



# Effects of $\text{CdCl}_2$ treatment on the local electronic properties of polycrystalline CdTe measured with PEEM

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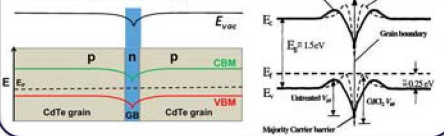
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## Introduction

Cadmium telluride (CdTe) is a polycrystalline photovoltaic (PV) material that has recently achieved power conversion efficiencies (21.5%) in research cells rivaling those of multicrystalline silicon. Further improvement of CdTe PV technology requires surpassing current limits in our understanding of fundamental mechanisms that influence processing-structure-property-performance relationships in CdTe. In particular, the influence of grain-boundary (GB) limitations and nano- or microscale nonuniformities have been identified as issues needing improved understanding.

**CdTe GB models** [Reviewed in J. D. Major, *Semicond. Sci. and Technol.* 31, 093001 (2016).]



## Objective & approach

- Objective: Determine processing-structure-property-performance-relationships critical for improving thin-film PV materials & devices

### Key Questions:

*How does  $\text{CdCl}_2$  treatment, a widely-used processing step, influence the local electronic structure of CdTe thin films?*

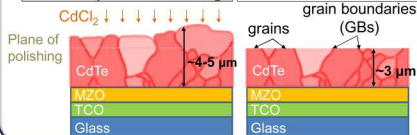
*Does oxygen (via air exposure) impact this step?*

## Sample preparation protocol

- NSG TEC 10 glass superstrates coated with transparent conducting oxide (TCO):  $\text{SnO}_2/\text{SiO}_2/\text{SnO}_2/\text{F}$
- 100 nm  $\text{Mg}_{0.23}\text{Zn}_{0.77}\text{O}$  (MZO) window layer
- 4-5  $\mu\text{m}$  closed-space sublimation (CSS) CdTe
- CdTe films are mechanically polished
- Low-energy sputtering step: 50 eV  $\text{Ar}^+$  ions / 10–20 min /  $\sim 0.1\text{--}0.15 \mu\text{A}/\text{cm}^2$
- Sample transfer with no additional air exposure by use of an inert gas (dry  $\text{N}_2$ ) glove box
- X-ray Photoelectron Spectroscopy was used to verify surface composition after preparation steps

### CdTe superstrate config.

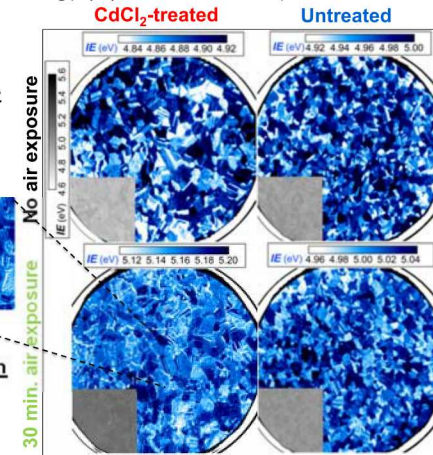
### Polished CdTe surface



## Local ionization energy (PEY)

Take a set of images while sequentially varying the wavelength ( $\lambda$ ). Extract an ionization energy (IE) value for each pixel.

Microcrystalline grain domains, twin boundaries, & polish marks are visible in all maps, as is **grain-to-grain variation** of the ionization energy.  $\text{CdCl}_2$  treatment **did not produce** grain boundary contrast.



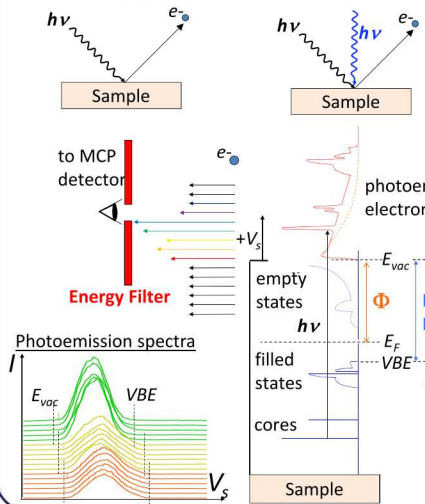
**$\text{CdCl}_2$  treatment and air exposure produced grain (GB) boundary contrast** consistent with scanning probe measurement of activated grain boundaries.

Oxygen increased the average ionization energy while  $\text{CdCl}_2$  treatment decreased the average ionization energy.

## Principles & methods: Spectroscopic PEEM

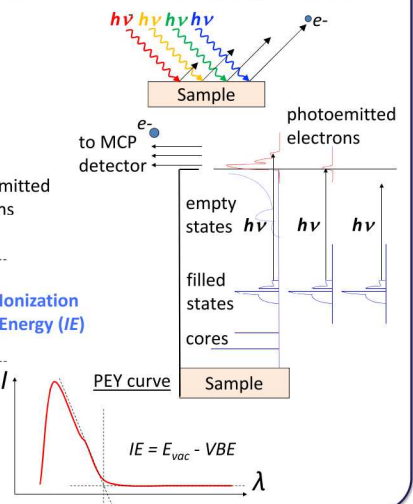
### Local Photoemission Spectroscopy (PES)

- Fixed energy, PES
- $\lambda = 190 \text{ nm}$ ,  $h\nu \sim 6.5 \text{ eV}$
- PES + Surface photovoltage (SPV)
- $403 \text{ nm}$ ,  $\sim 3 \text{ mW}$  CW laser



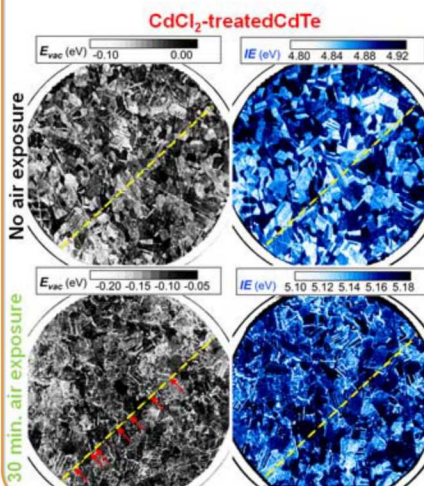
### Local Photoemission Yield (PEY)

- Variable energy, PEY
- $\lambda = 175\text{--}350 \text{ nm}$ ,  $h\nu = 3.6\text{--}7 \text{ eV}$

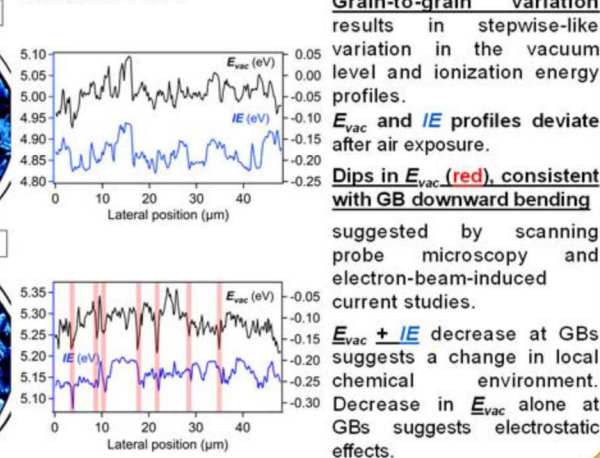


## Local photoemission spectra (PES)

Take a set of images (stack) while sequentially varying the start voltage ( $V_s$ ). Extract vacuum level cutoff ( $E_{vac}$ ) threshold at each pixel.



**Grain-to-grain variation** of the vacuum level indicates work function variation. Vacuum level and ionization energy profiles track one another.



**Grain-to-grain variation** results in stepwise-like variation in the vacuum level and ionization energy profiles.

$E_{vac}$  and IE profiles deviate after air exposure.

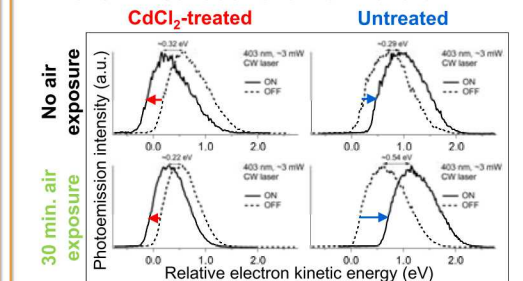
**Dips in  $E_{vac}$  (red), consistent with GB downward bending**

suggested by scanning probe microscopy and electron-beam-induced current studies.

$E_{vac} \pm IE$  decrease at GBs suggests a change in local chemical environment. Decrease in  $E_{vac}$  alone at GBs suggests electrostatic effects.

## Surface photovoltage measurements

With and without additional illumination.



Additional illumination shifted  $E_{vac}$  to **lower energy** for  $\text{CdCl}_2$ -treated CdTe  $\Rightarrow$  **upward** surface band bending, while  $E_{vac}$  shifted to **higher energy** for untreated CdTe, indicating **downward** surface band bending.

The general model of SPV interprets the switch from **downward** to **upward** surface band bending as inversion from **p-type** to **n-type**.

Oxygen decreased/increased  $E_{vac}$  shifts for  $\text{CdCl}_2$ -treated/untreated CdTe. Along with ionization energy trends, this suggests that oxygen may passivate CdTe defects or perhaps act as a hole dopant.

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