

# Electrical and Optical Characterization of Ultrathin Tellurium Nanostructures Synthesized by Vapor Phase Deposition

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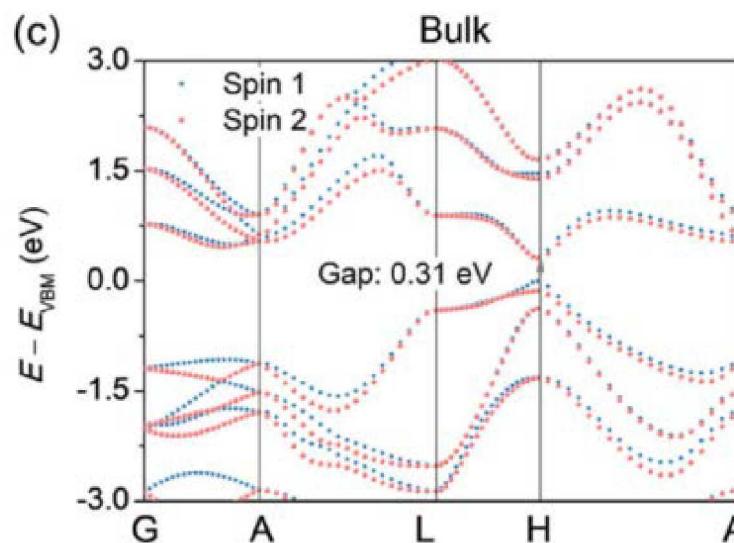
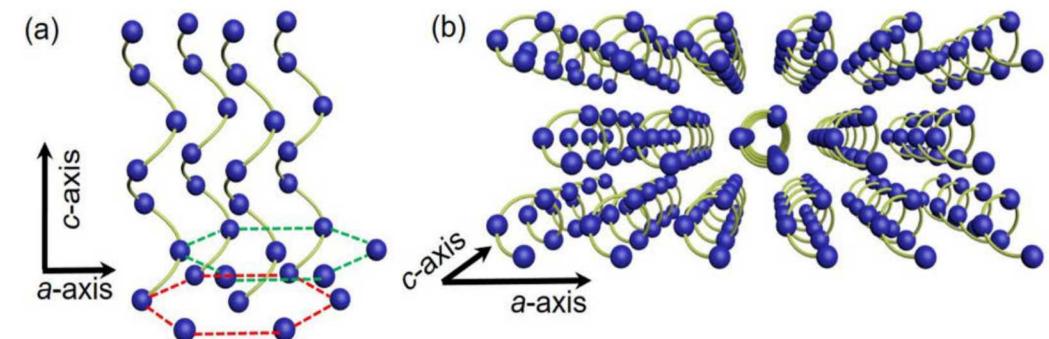
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# Outline

- Motivation: Te from bulk to layered structures
- Previous synthesis approaches
- High temperature synthesis of ultrathin Te nanostructures
- Structural, electrical, and optical characterization
- Conclusions

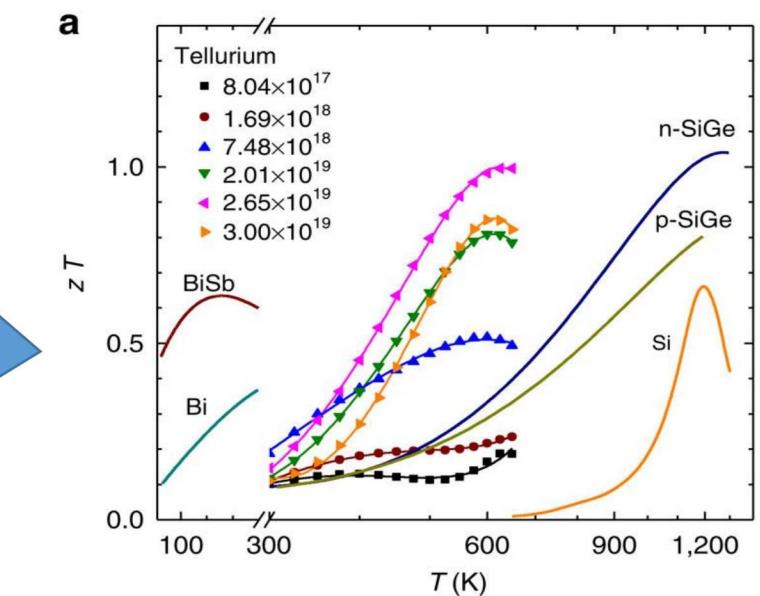
# Bulk Tellurium

- 1D helical chains of Te atoms staked together on 2D hexagonal plane
- Covalent bond between neighboring atoms in the same chain
- van der Waals type bond between neighboring atoms across the chain



Nearly direct bulk band gap of 0.33 eV

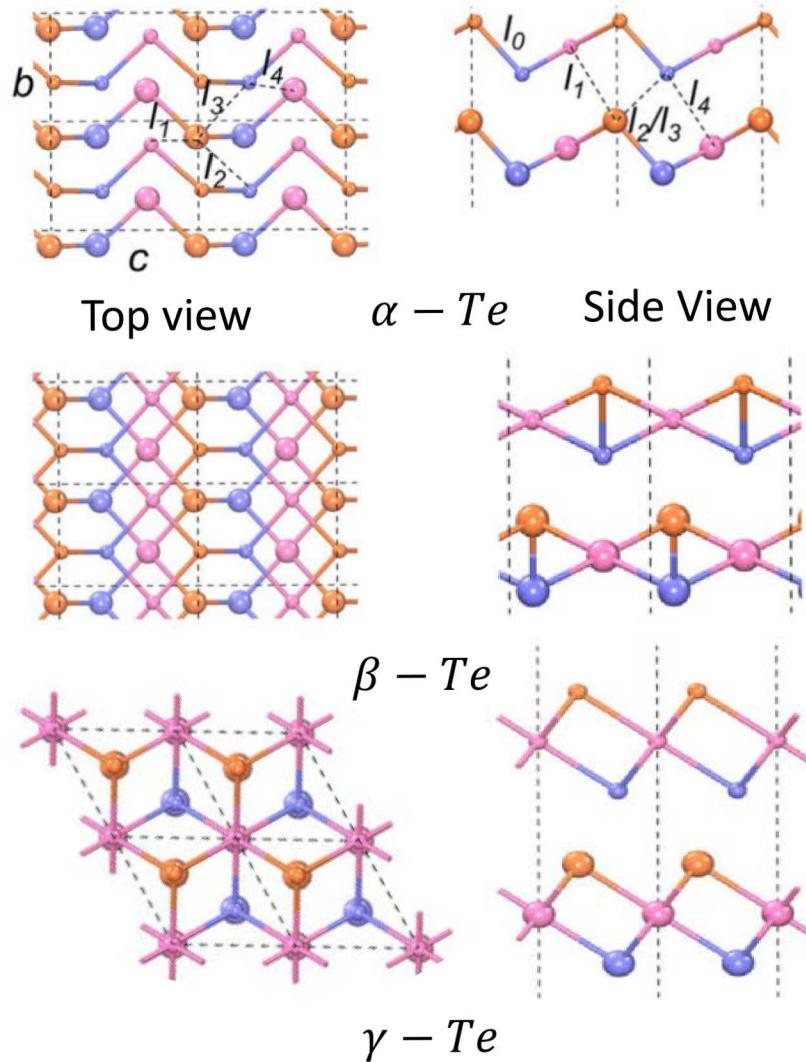
- Te exhibits high thermoelectric performance in bulk
- Several Te based compounds are excellent thermoelectric materials e.g. PbTe



J. Qiao et al. Science Bulletin 63 (2018) 159–168

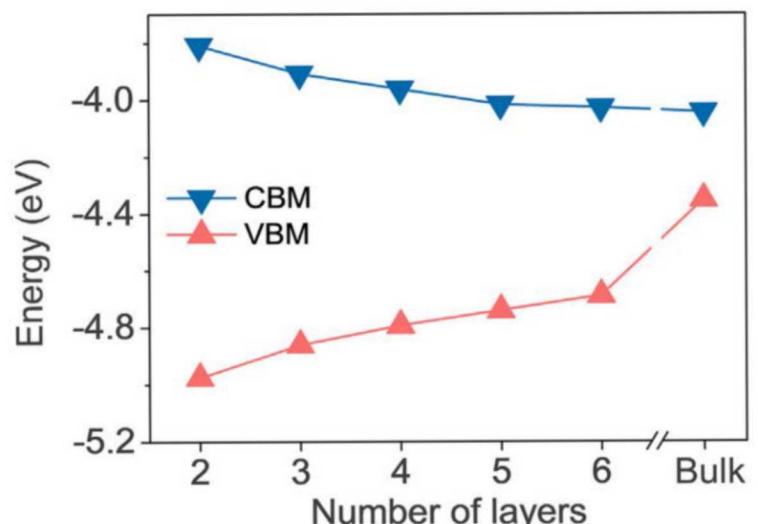
Lin, S. et al. Nat. Commun. 7:10287 10287 (2016).

# Layered Structure of Tellurium

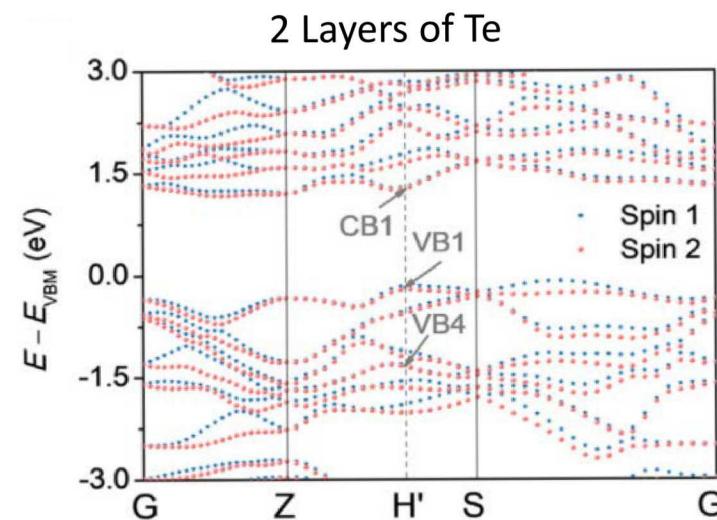


- Unique crystal structure allows to synthesize Te in 1D and 2D form
- 2D tellurium is equivalent to transition metal dichalcogenides of formula  $MX_2$  (e.g.  $MoS_2$ ) where M is replaced by Te.

Prediction\*: 2D structure of Te (Tellurane) exists in 3 phases:  $\alpha$ -,  $\beta$ -, and  $\gamma$ -Te



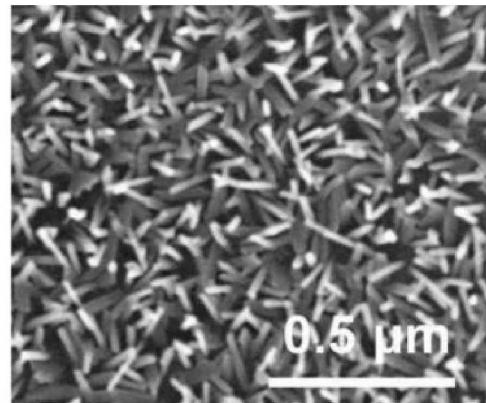
Tunable bandgap with Te thickness



\*J. Qiao et al. Science Bulletin 63 (2018) 159–168

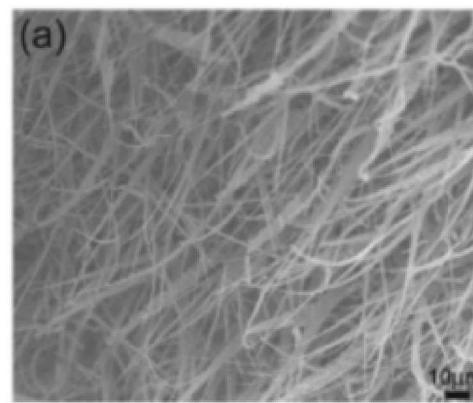
High hole mobility predicted along surface:  $\sim 10^5 \text{ cm}^2/\text{V.s}$

# Bottom-up and Top Down Synthesis of Te Nanostructures



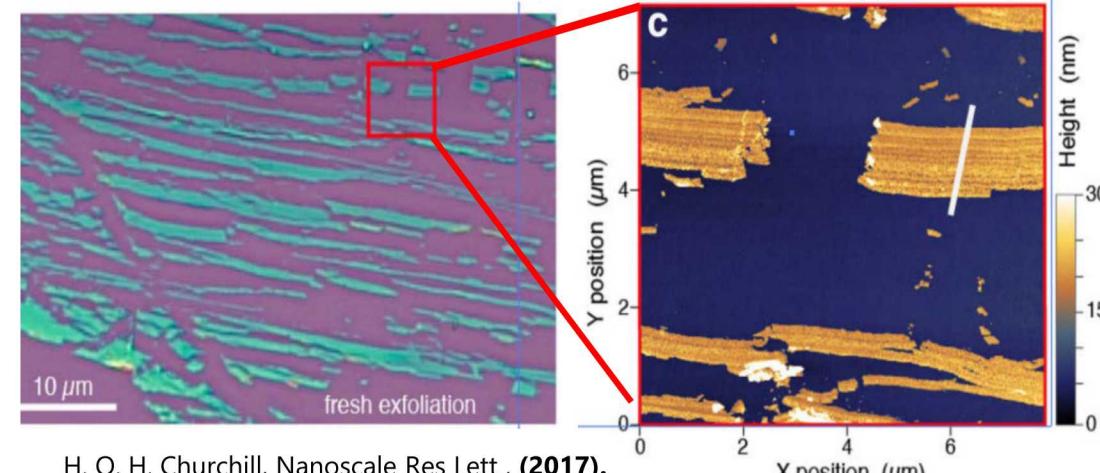
G. Zhou, Adv Mater, (2018).

MBE deposited nanostructures



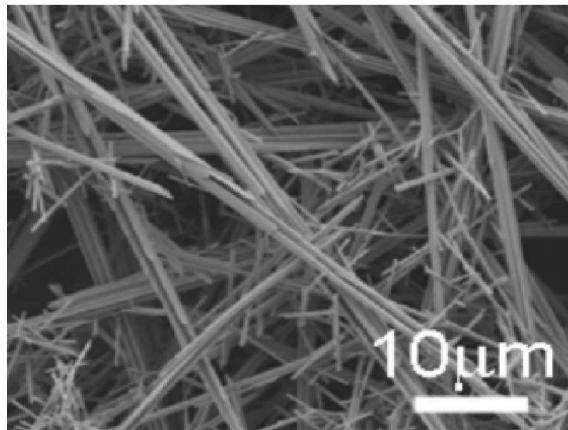
Q. Wang, J. Phys. Chem, (2007).

Vapor phase deposition at 100°C



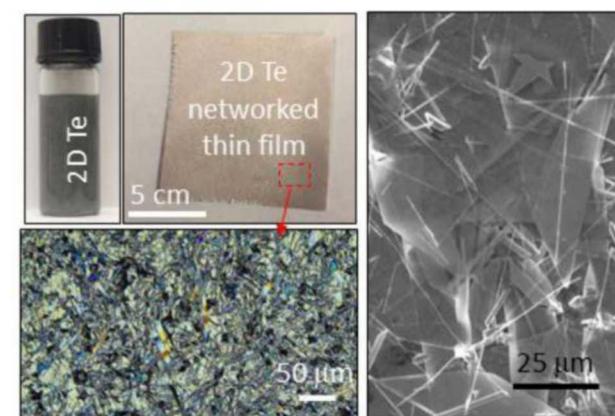
H. O. H. Churchill, Nanoscale Res Lett, (2017).

Exfoliation



J. M. Song, Cryst Growth, (2008).

Solvothermal synthesis using TeO<sub>2</sub>



Y. Wang, Nature Electronics, 1 (4), 228-236 (2018).

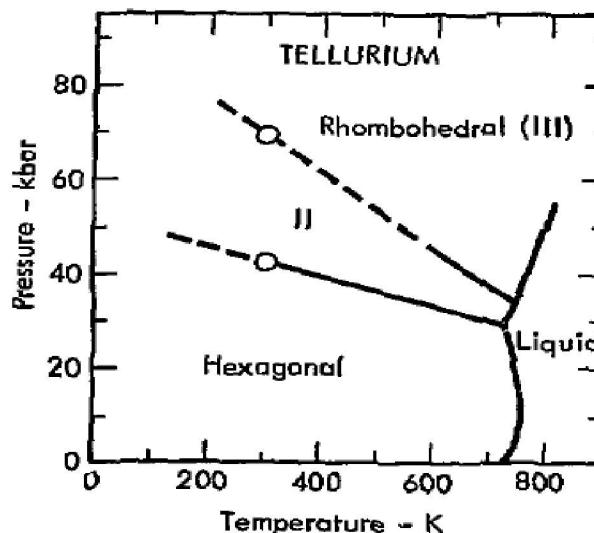
Solution based synthesis using Na<sub>2</sub>TeO<sub>3</sub>

# High Temperature Synthesis of Ultrathin Te Nanostructures

## Challenges to synthesize high quality Te nanostructures by conventional methods

- Solution based method usually contaminates the nanostructures by chemical byproducts
- Low temperature MBE/ Vapor phase deposition is susceptible to crystal defects and low quality growth

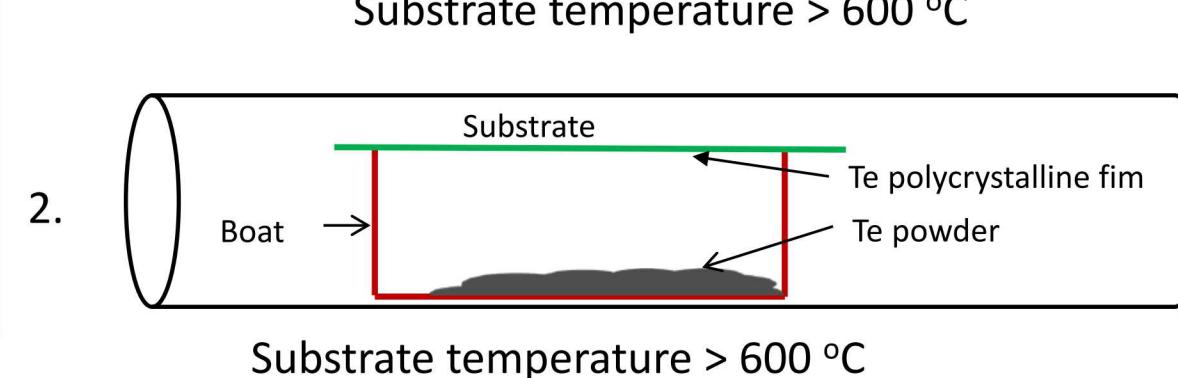
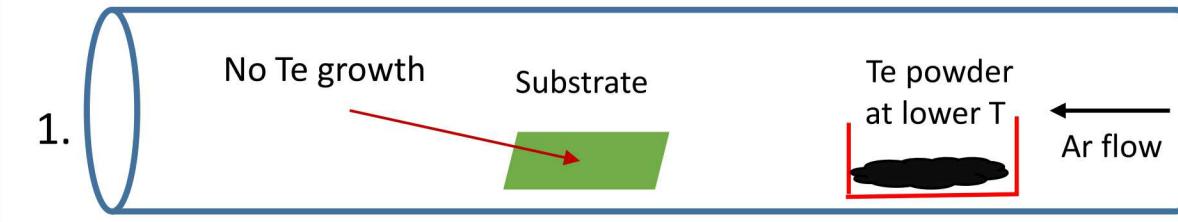
High temperature vapor phase deposition is the desired method to produce high quality Te nanostructures



## Major Challenge

Te has high vapor pressure/evaporation rate:  
difficult to control Te deposition and re-evaporation at high substrate temperature

## Attempts to grow high temperature Te nanostructures



# High Temperature Synthesis of Ultrathin Te Nanostructures

## Overcoming challenges

### ZrTe<sub>2</sub> powder as the Tellurium source

ZrTe<sub>2</sub> decomposes slowly at > 450 °C into crystalline Zr and Te gas\*

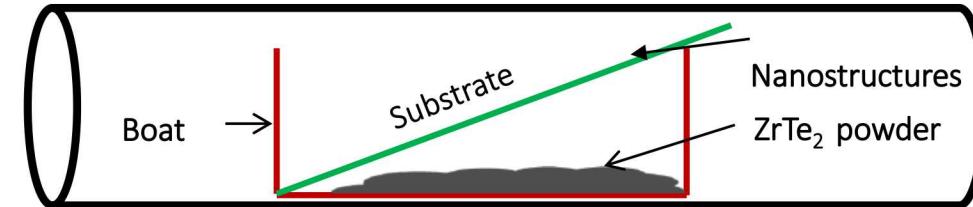


\*G.K. Johnson, 17 (1985)

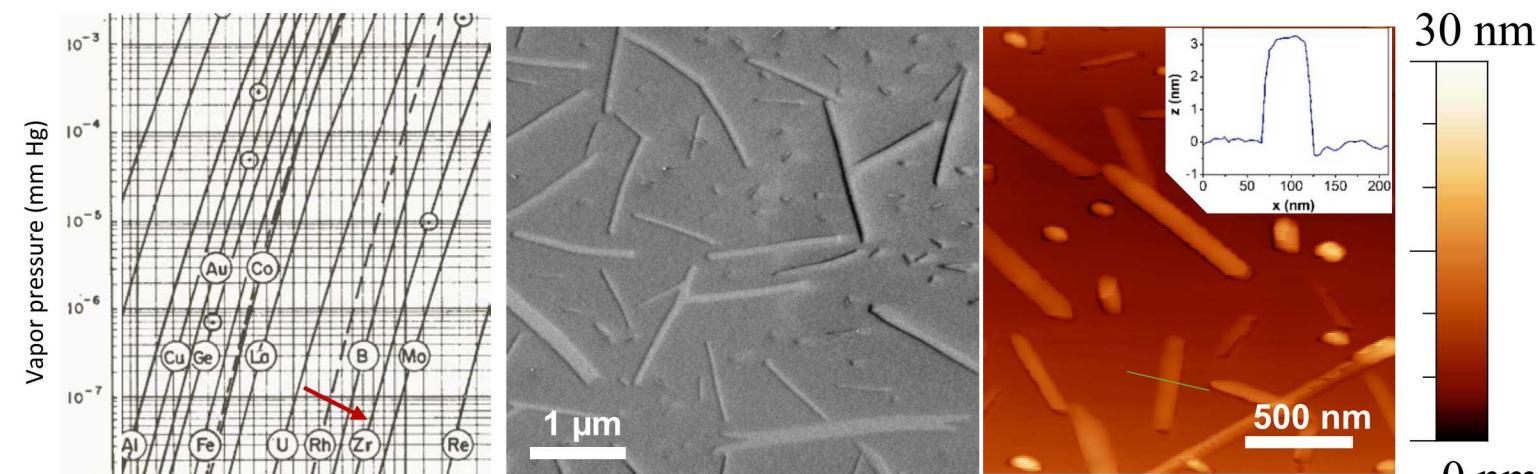
Crystalline Zr evaporate only above 1500 °C

Slow decomposition of ZrTe<sub>2</sub> controls the Te vapor pressure at high temperature and containment of Te evaporation by covering with substrate successfully grew Te nanostructures at > 600 °C .

Te nanostructures thickness: down to 3 nm  
Width and length: few hundred nm and few  $\mu\text{m}$



Growth temperature ( $T_g$ ):  $600 \leq T_g \leq 750 \text{ } ^\circ\text{C}$



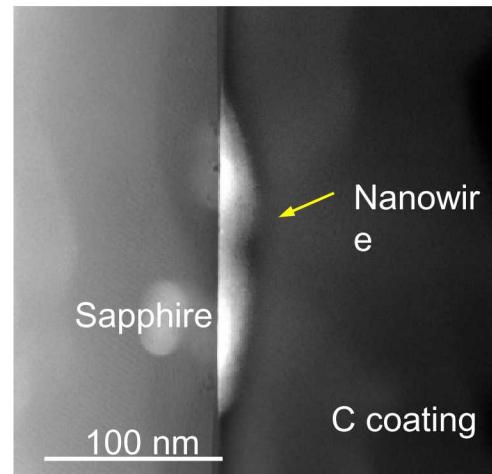
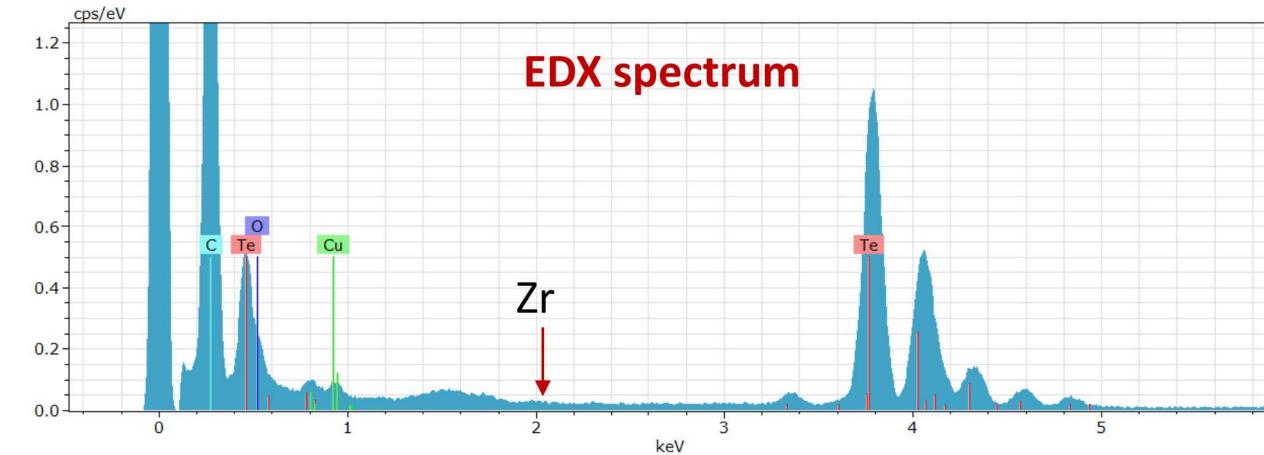
SEM image

AFM image

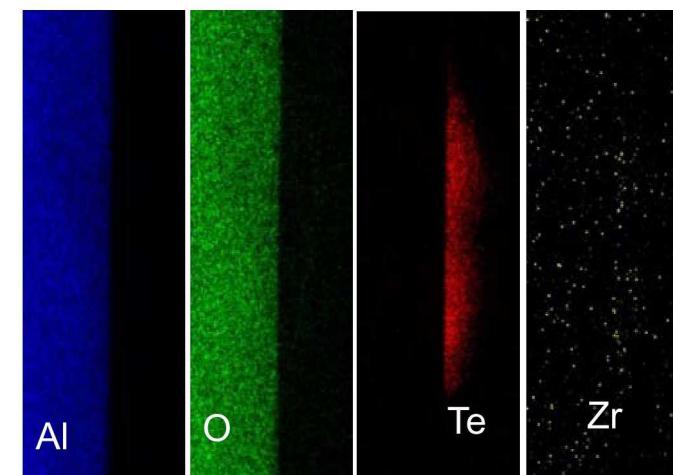
# Structural Characterizations

Source material  $\text{ZrTe}_2$  → Are these nanostructures pure Te?

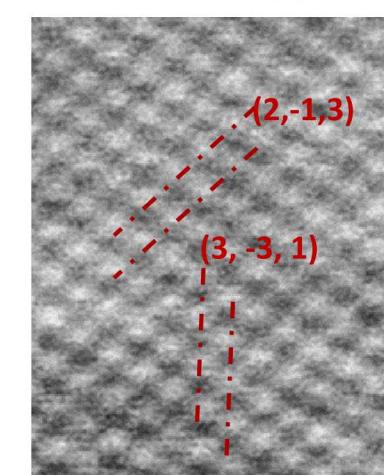
- Te nanostructures are studied under STEM
- Electron Dispersive X-ray studies could not reveal any presence of Zr on the nanostructures to the detection limit of instrument
- **Nanostructures are pure Te**



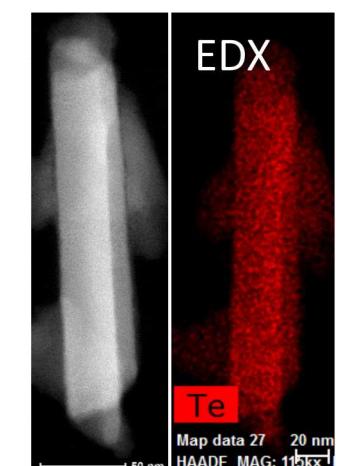
Cross-sectional STEM image of Te Nanowire



Elemental mapping by EDX



HRTEM image of the Te cross section

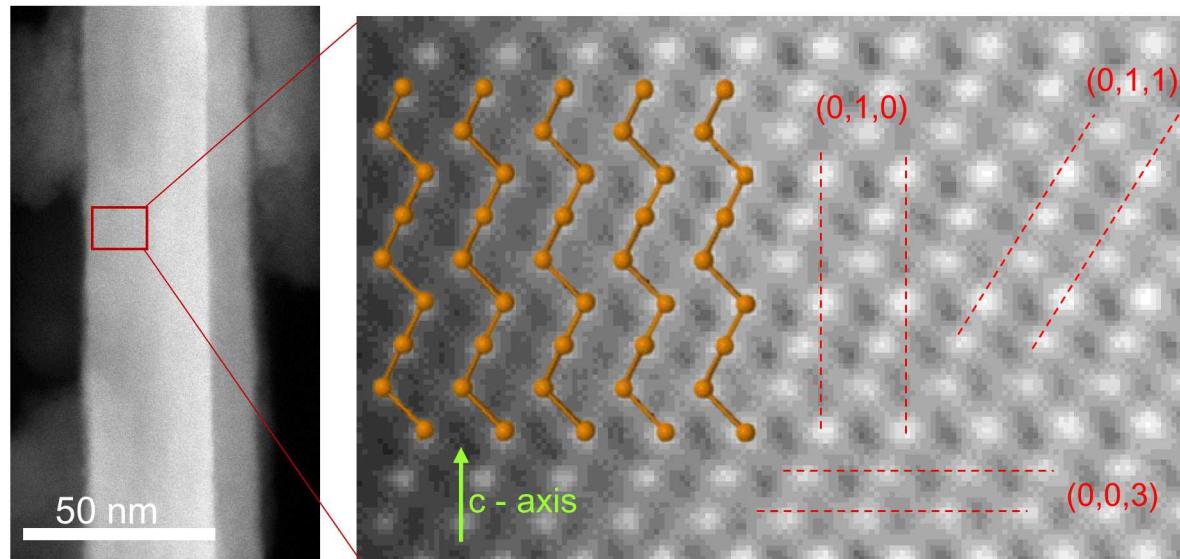


Top view

# Structural Characterizations

## HAADF and FFT : Te crystal structure matches with $\alpha$ – Te phase

- High-angle annular dark-field (HAADF) STEM image exhibits helical chains of Te
- **Trigonal crystal structure with hexagonal cell**
- Space group  $P3_121$
- Lattice parameters:  $a = 4.458 \text{ \AA}$ , and  $c = 5.927 \text{ \AA}$

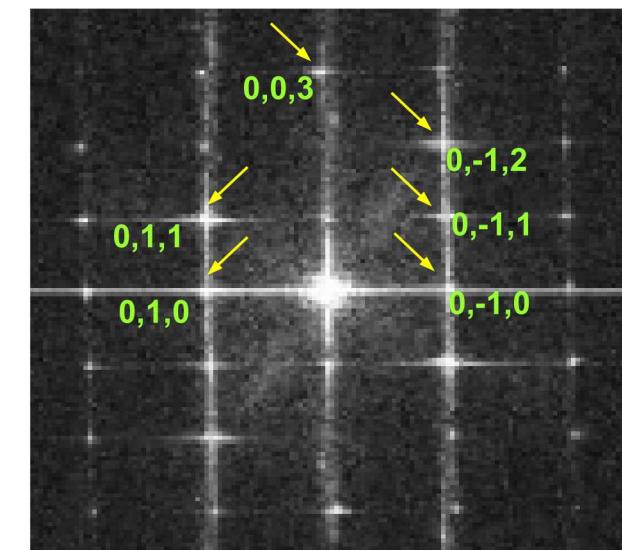


Te nanowire

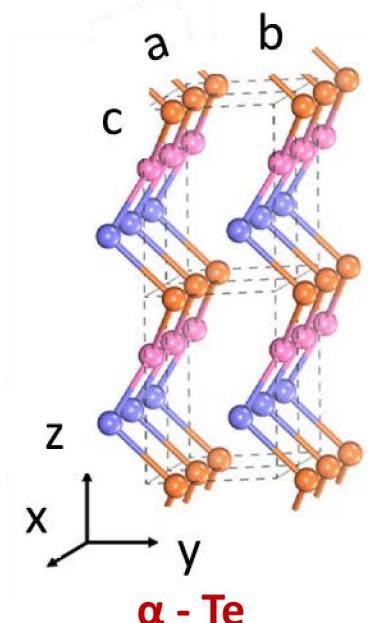
High-angle annular dark-field (HAADF) STEM image

## FFT analysis

Miller index	D-spacing		
	Expected (Å)	Measured (Å)	Difference (%)
0,1,0	3.856	3.83	-0.67
0,1,1	3.233	3.2	-1.02
0,-1,2	2.35	2.32	-1.28
0,0,3	1.975	1.94	-1.77



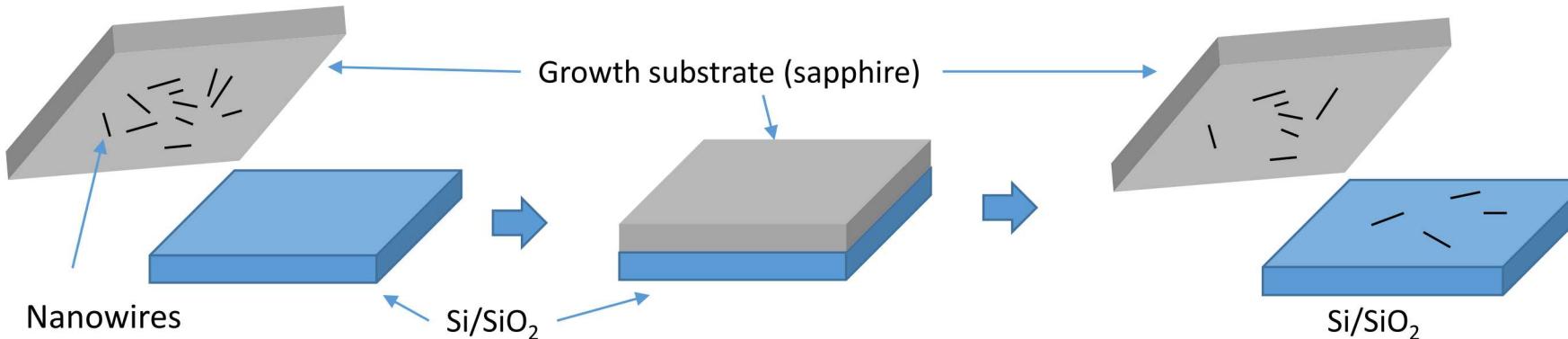
FFT patterns



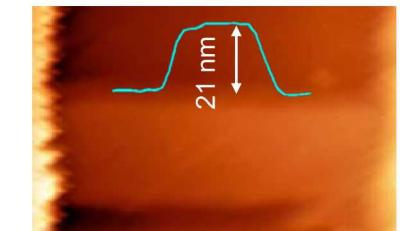
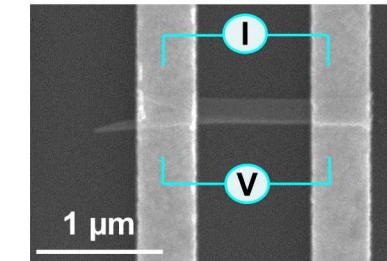
# Electrical Properties of a Te Nanostructure

## Device fabrication and measurement

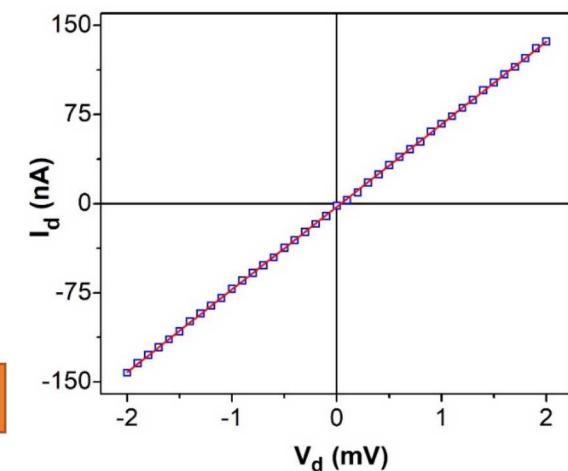
- Challenging to fabricate gated device on the growth substrate (sapphire)
- Nanostructures were transferred to  $\text{Si}/\text{SiO}_2$  substrate by contact method
- Electron beam lithography was carried out to fabricate the single nanostructure devices
- In-situ ion milling at the contact region before metallization to improve contacts
- Nanowire channel length = 830 nm and width = 180 nm, thickness = 21 nm
- I-V characteristics is linear at room temperature  $\rightarrow$  no Schottky barrier observed



Nanowire transfer by contact method. To enhance the transfer, growth substrate was soaked with DI water



AFM image of nanowire device



# Electrical Properties of a Te Nanostructure

**Resistivity ( $\rho$ ) = 645  $\mu\Omega\cdot\text{cm}$ .**

- Nearly two order of magnitude lower than the resistivity of bulk-Te
- Significantly lower than solution-based synthesized Te nanostructures

**Gated measurements at room temperature**

- Negative transconductance ( $g_m$ ): ***p*-type semiconductor**

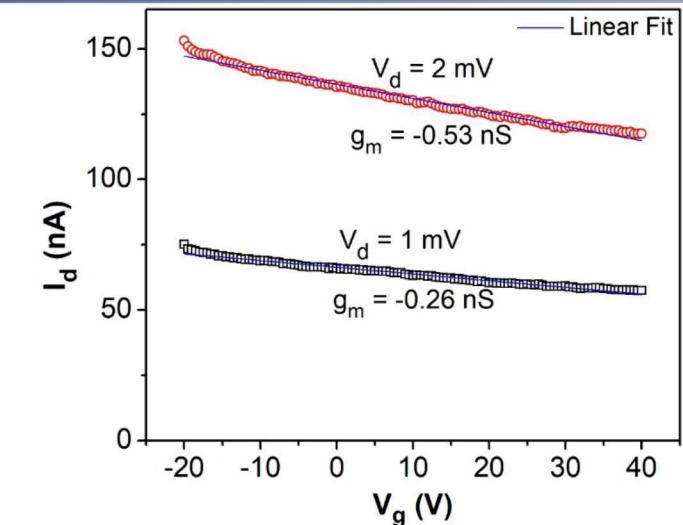
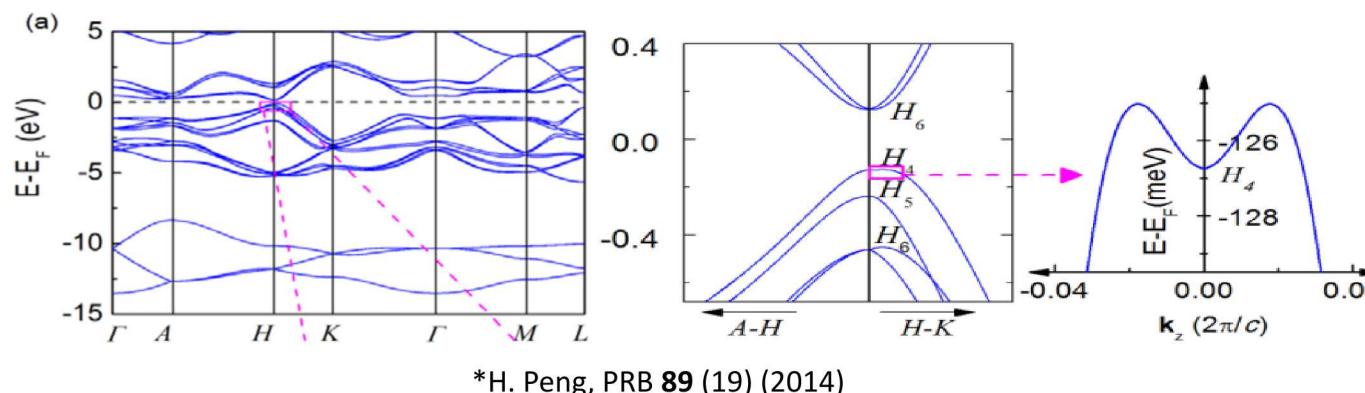
$$g_m = d I_d / d V_g \quad \mu_h = \frac{g_m L^2}{V_{ds} C_{get}}$$

$$C_{get} = \epsilon_0 \epsilon_r w L / t$$

- Hole mobility ( $\mu_h$ ) = 349  $\text{cm}^2/\text{V}\cdot\text{s}$
- Mobility higher than of  $\text{MoS}_2$  which is typically  $\sim 190 \text{ cm}^2/\text{V}\cdot\text{s}$  (L. Ma, APL **105** (7) (2014))

$$n_h = 1/e \rho \mu_h$$

- Hole concentration ( $n_h$ ) =  $2.78 \times 10^{18} \text{ cm}^{-3}$

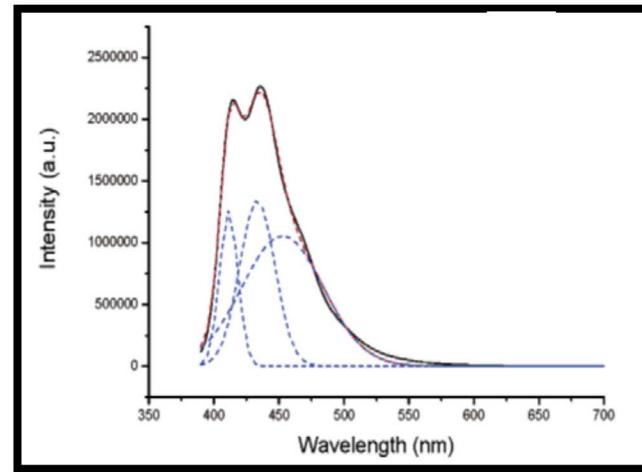


## Hole Transport Mechanism\*

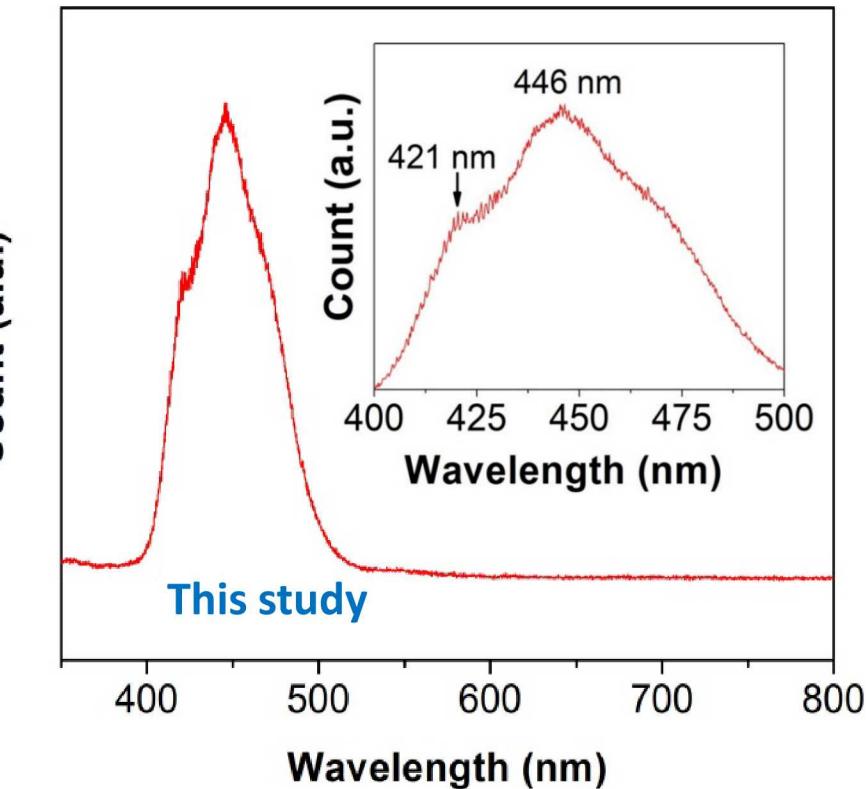
- Unique valence band structure at the H point in the Brillouin zone provides conduction channels for the holes
- The four fold degenerate valence band at H point is split into two non-degenerated  $H_4$  and  $H_5$  bands and a doubly degenerated lower  $H_6$  band due to the strong spin-orbit coupling in the Te.
- Only  $H_4$  and  $H_5$  bands lie close to the Fermi level and thus contribute holes transport.

# Micro-PL study

- Room-temperature micro- photoluminescence (micro-PL) measurements show a strong violet-blue luminescence at  $\sim 445$  nm
- PL peak lies at a significantly higher energy level than the expected bandgap level.



Previously observed PL in Te nanowires  
*H. Sheng, Langmuir, 22 (2006)*



- Hartee-Fock-Slater model:
  - Peaks in the energy range of 0-3 eV can be assigned to the transition form valence band p-bonding (VB3) to conduction band p-antibonding (CB1).\*

\**T. Ikari, Mater. Res. Bull. 21, 99 (1986)*

# Conclusions

- Te can be synthesized in 2D form which can exists in 3 phases:  $\alpha$ -,  $\beta$ -, and  $\gamma$ -Te
- 2D Te is equivalent to transition metal dichalcogenides of formula  $MX_2$  (e.g.  $MoS_2$ ) where M is replaced by Te
- High temperature vapor phase deposition of Te nanostructures was realized by using  $ZrTe_2$  as Te source
- Ultrathin nanostructures were obtained down to 3 nm thickness
- The synthesized Te nanostructures exhibited  $\alpha$  – Te phase
- The high quality Te nanostructures exhibited 2 order lower resistivity that than of bulk Te and chemically synthesized Te nanostructures
- High hole mobility was observed ( $349 \text{ cm}^2/\text{V.s}$ ) which is greater than for typical 2D van der Waal materials
- Micro-PL exhibits luminescence at  $\sim 445 \text{ nm}$  which is significantly deeper energy level than expected bandgap