

# Short term creep of P91 in CO<sub>2</sub> environment utilizing confocal microscopy

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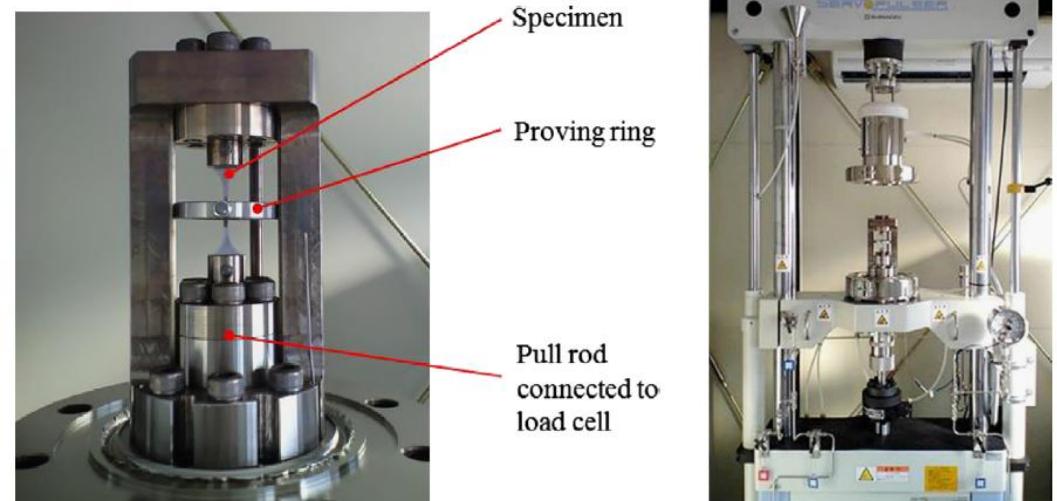
# Need for In-situ Strain Measurements

- Problem: Difficult to measure strain in controlled environments at high temperature or pressure
- Existing environmental chambers have limited strain observation capabilities
  - Baffles prevent strain observation and spring rate must be accounted for when measuring load, I.E. Kubota et al.
  - Inferred from load cages, I.E. Saengas et al.
- Commercial systems not sealed, limited to 450-550°C, (MTS)
- Typical In-situ strain observations are done in SEM under vacuum or low vacuum I.E. environmental SEM

High pressure hydrogen chamber – Kubota et al.

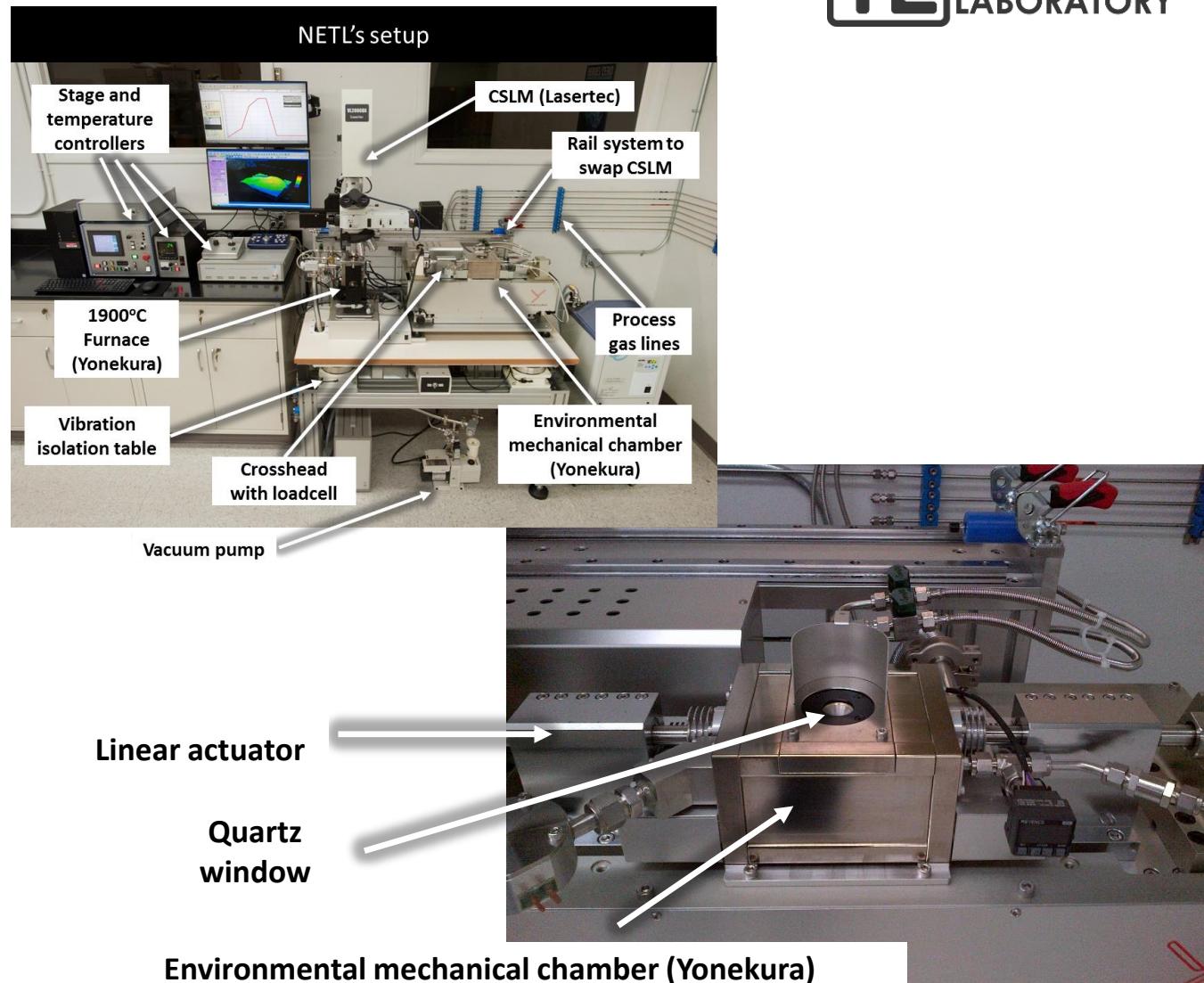


Elevated temperature / pressure autoclave – Saengas et al.



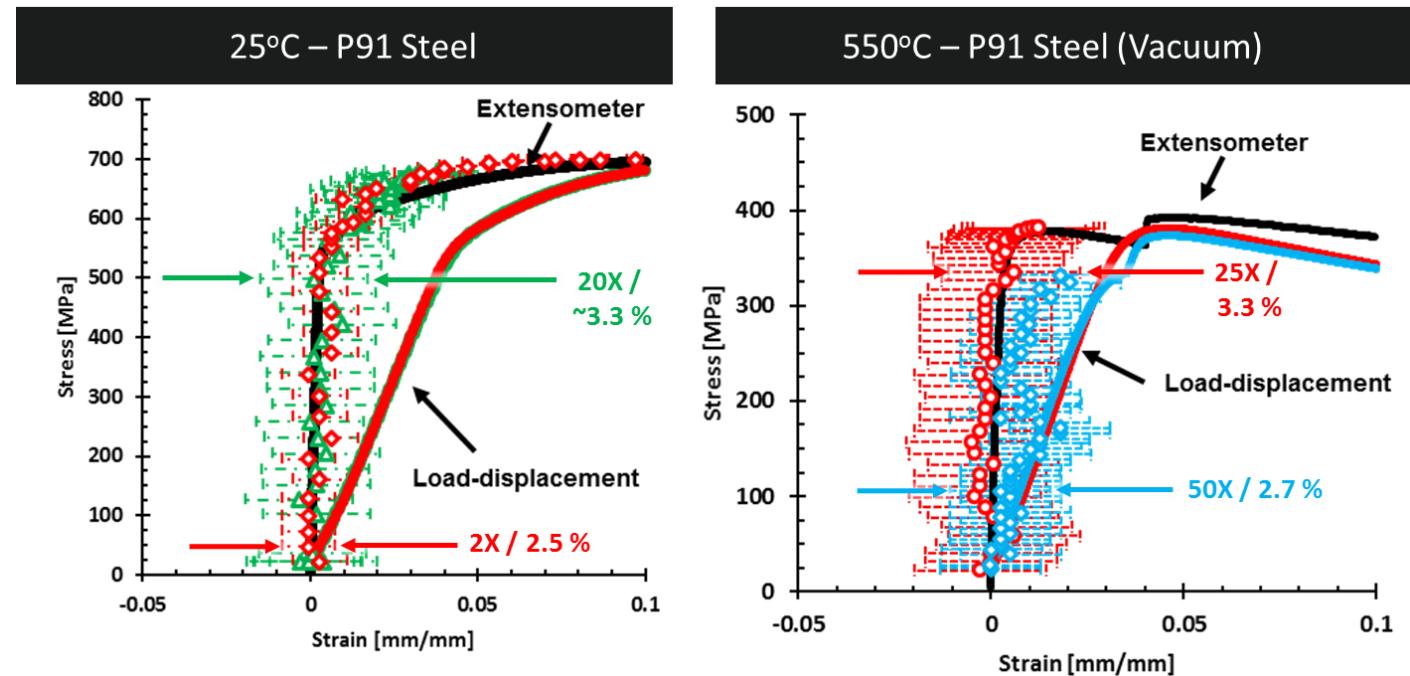
# Confocal Scanning Laser Microscope (CSLM)

- CSLM mounted on rail system
  - Allows for easy access to both very high-temperature stage and mechanical stage
  - 405 nm laser
  - 2X to 10X objective lens with digital zoom capability
- Load stage inside sealed environmental mechanical chamber with quartz viewing window
  - Process gas or vacuum
  - Heating by focusing dual halogen lamps
  - 5000 N load stage
  - Max strain rate 20 mm/min
- CSLM allows for strain measurement at high temperatures in controlled environments



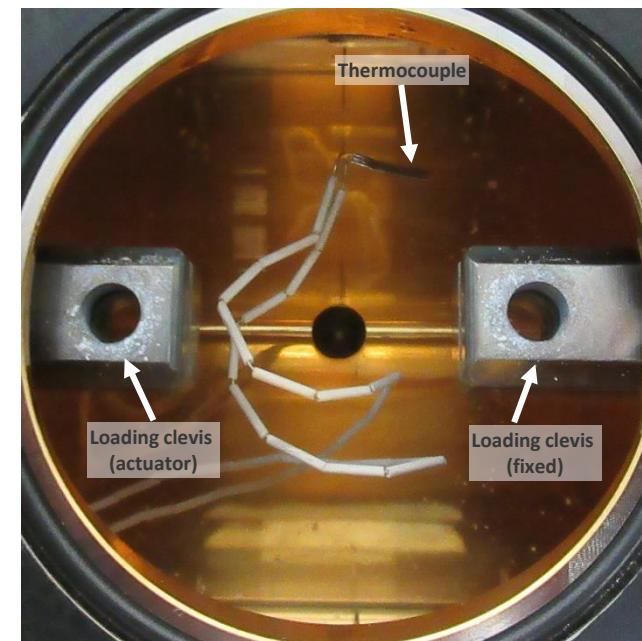
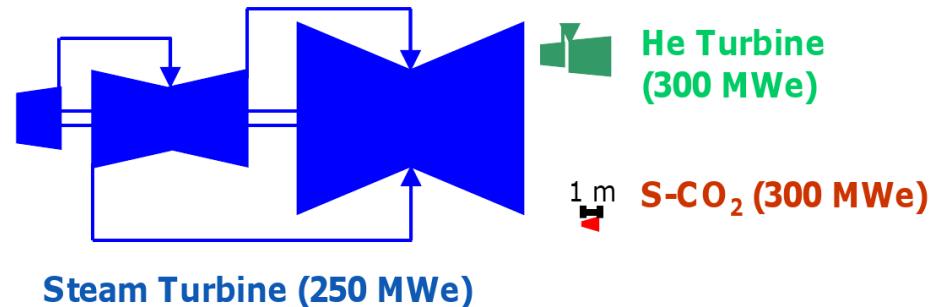
# Previous Results

- Tested tensile properties of P91 steel in Air at 25°C, Vacuum at 550°C and CO<sub>2</sub> at 550°C
- Tensile curves overlapped with ASTM E8 generated data
- Greater experimental uncertainty at higher magnifications
- Noted image wobble was likely causing uncertainty
- Could be improved with automated strain measurements



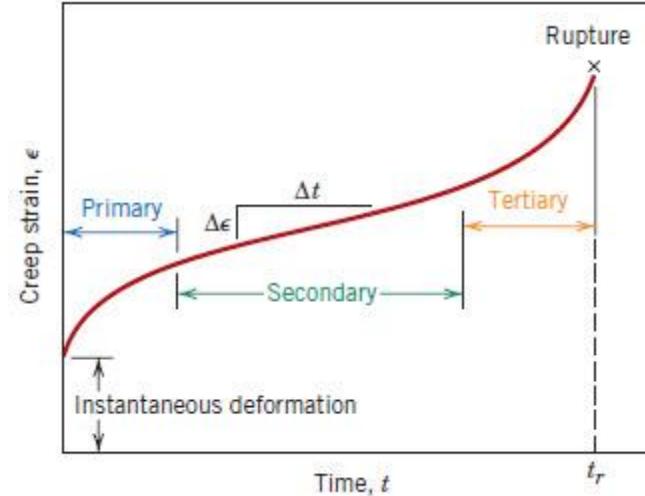
# Creep Testing

- Building upon success of tensile tests
- Utilize system for *in-situ* controlled environment creep testing of materials
- Note corrosion properties pressure dependent for steels [Holcomb et al.]
- Test and analyze creep at 550°C under
  - CO<sub>2</sub> – Of importance to NETL advanced combustions projects
  - Ar – Simulates vacuum, -50 kPa-g was largest vacuum we could pull
- Target short term creep tests
  - 10 h
  - 20 h
  - 100 h
- Goal: Observe how CO<sub>2</sub> effects creep rates of P91 steel



# Creep Background

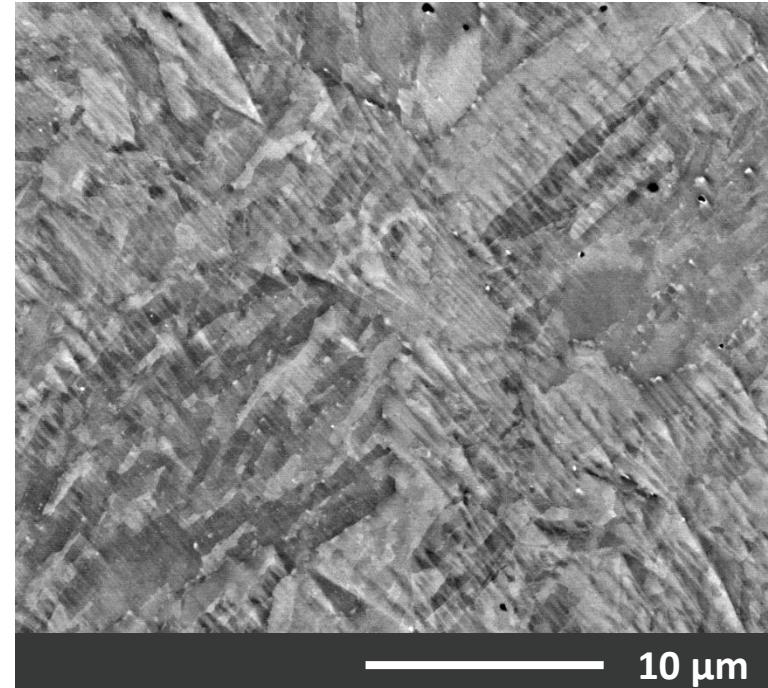
- When  $T/T_{melt} > 0.4$ , metals will elongate under load
- Strain rupture plots typically have three zones
  - Initial creep – rapid decrease in strain rate
  - Steady state creep – constant creep rate
  - Tertiary creep – rapid increase in creep rate
- Multiple models, Typically fit with exponential and Arrhenius fits
  - $\dot{\epsilon}_{ss} = K \cdot \sigma^n \cdot \exp(-Q/RT)$
- Larson-Miller parameter (LMP)
  - Hardness was observed to follow a Time-Temperature relationship
  - Observed to be valid for creep as well
  - $LMP = T \cdot (\log t_r + C)$
- Monkman-Grant
  - Relationship between minimum creep rate and rupture time
  - $t_r \cdot \dot{\epsilon}_{ss} = \text{Constant}$



- P91 steel from Arcelormittal Plate LLC

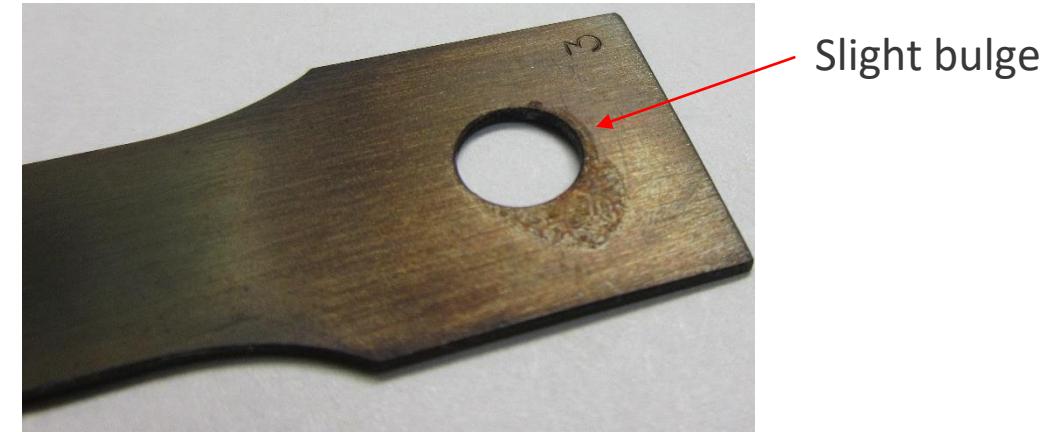
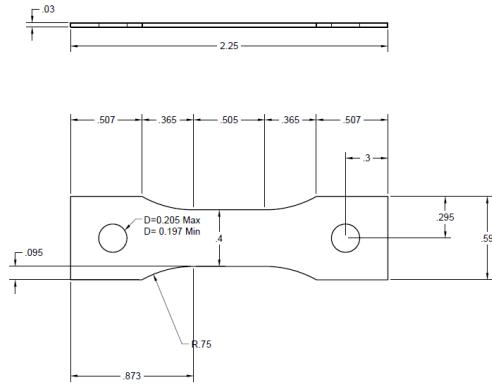
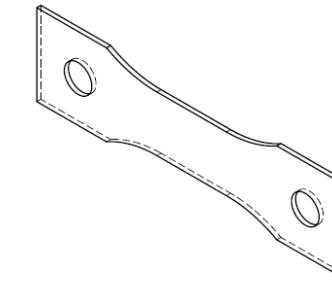
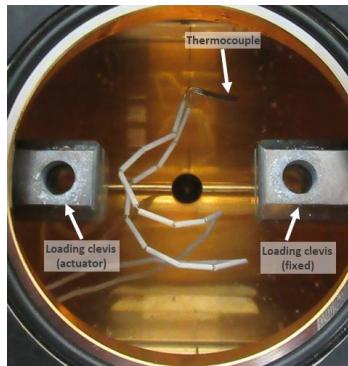
Element	Fe	C	Mn	P	S	Cu	Si	Ni	Cr	Mo	V	Ti	B	Al	Zr	Cb	N
Wt%	bal	0.09	0.45	0.01	0.004	0.09	0.33	0.09	8.37	0.9	0.224	0.002	0.001	0.007	0.001	0.072	0.0452

- Tempered martensite microstructure
- Small carbides line ferrite grain boundaries



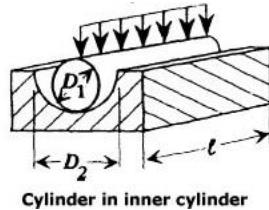
# Utilizing Dog-bone Specimens

- CLSM equipped with pinned and threaded clevises
- Originally used thin dog-bone specimens
- Maximize surface area to enhance environmental interactions with surface
  - $0.76 \times 10$  mm gage cross sectional area
- Specimens bulged at loading pin interface
- Tested at 190 MPa / 550C, Ar flow (50 SCCM)
  - Note 1800 N is known to buckle for this loading pin geometry
  - 190 MPa corresponds to 1100 N for this geometry

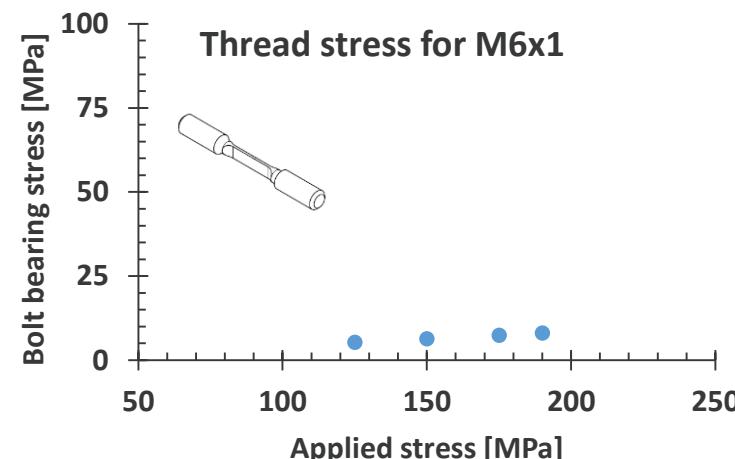
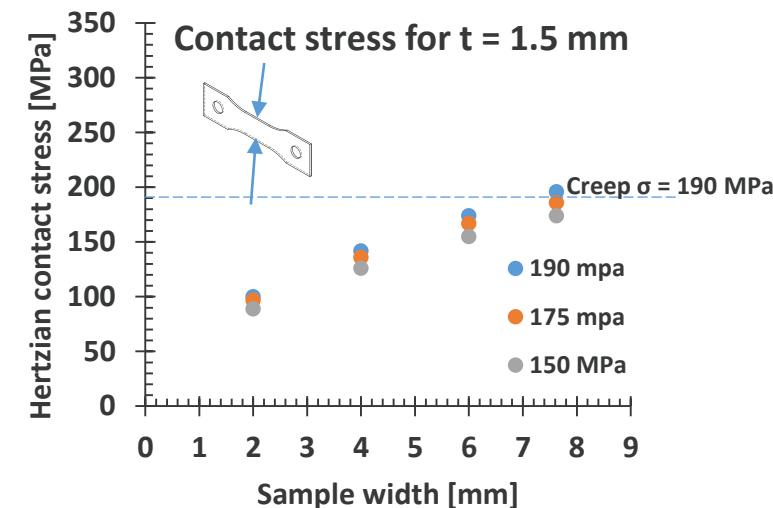
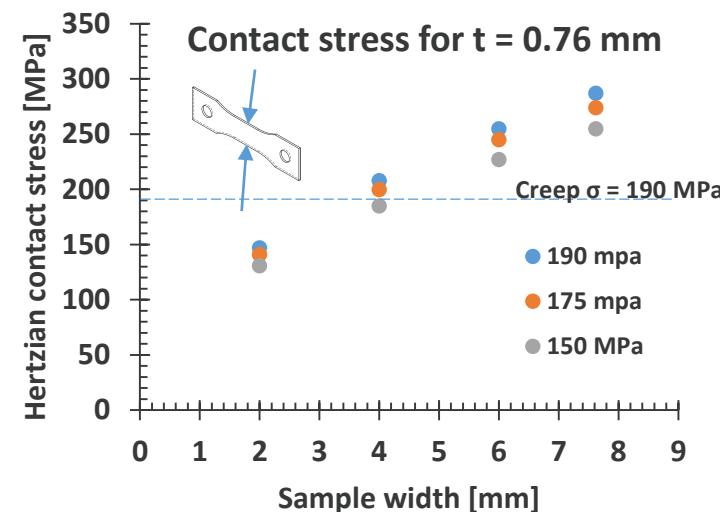


# Hertzian Contact Stress

- Looked at Hertzian contact stress for cylinder in side cylinder

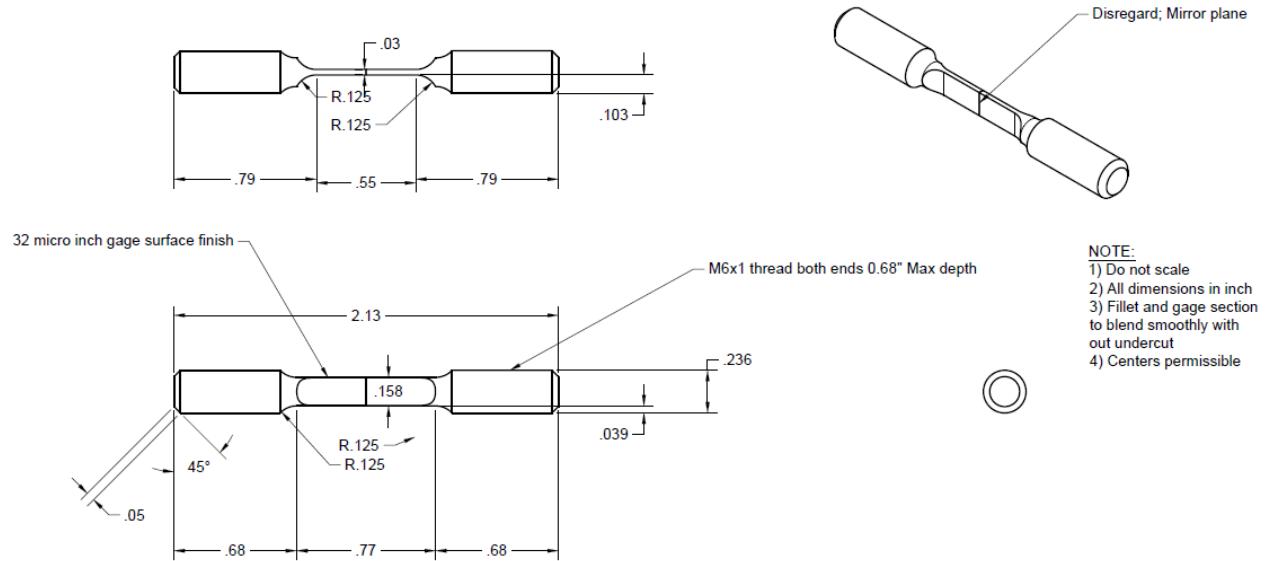


- Contact stress was always 0.5x applied stress or greater for dog-bone specimens even when increasing thickness
- Redesign to utilize threaded ends
- Thread stress  $< 0.10x$  applied stress



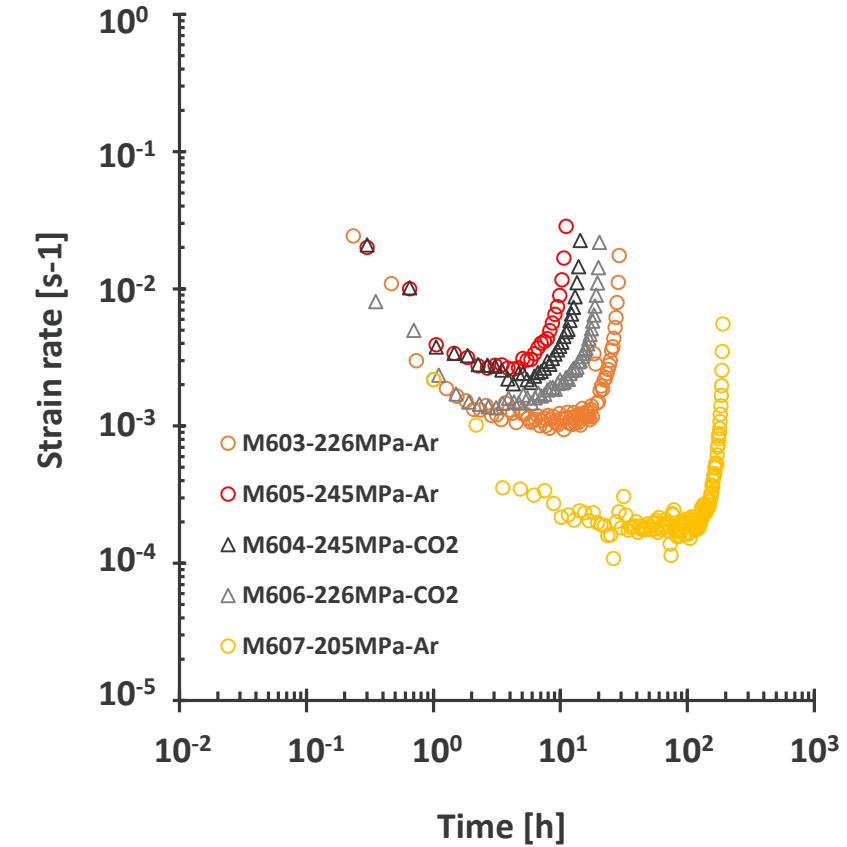
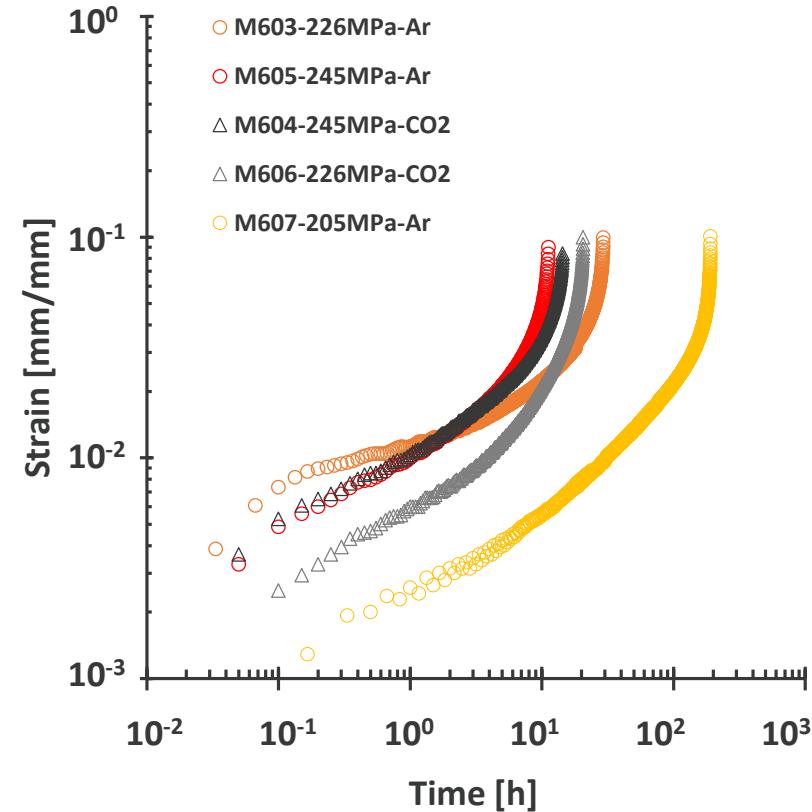
# Experimental Parameters

- Utilized 'compression sample'
- Modified to reduce gage thickness
- Maximizes surface area
- Machined from  $\frac{1}{2}$ " plate, L-T orientation,  $1/8$ " away from edges
- Test at  $550^{\circ}\text{C}$ 
  - Ar flow, 50 SCCM
  - $\text{CO}_2$  flow, 50 SCCM
- Creep at:
  - 245, 226, 205 MPa
  - Short term to evaluate system performance



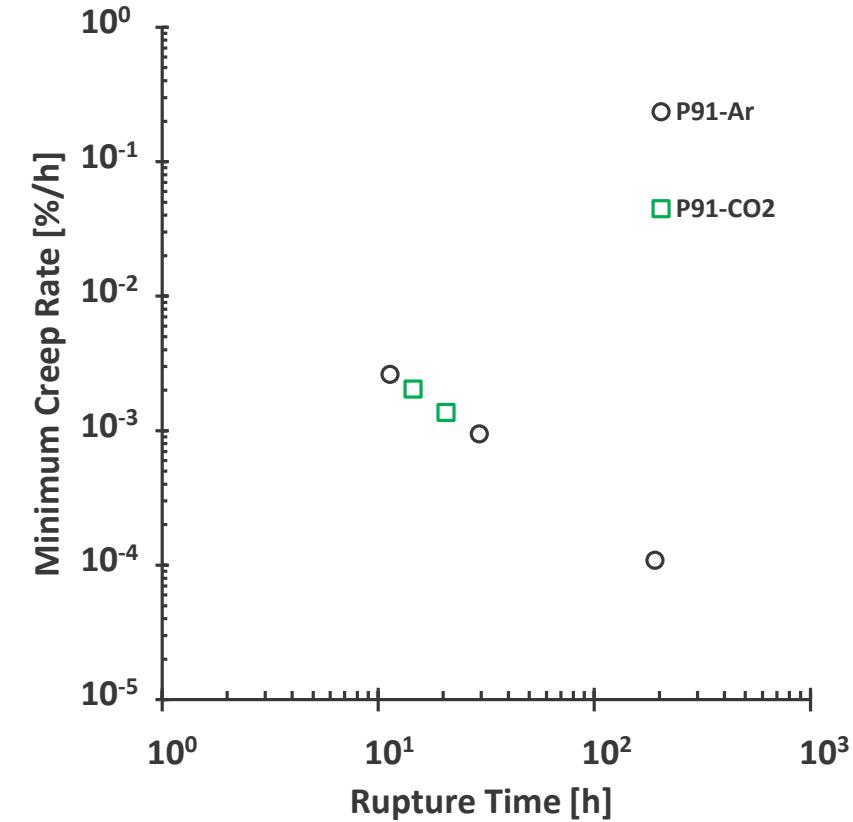
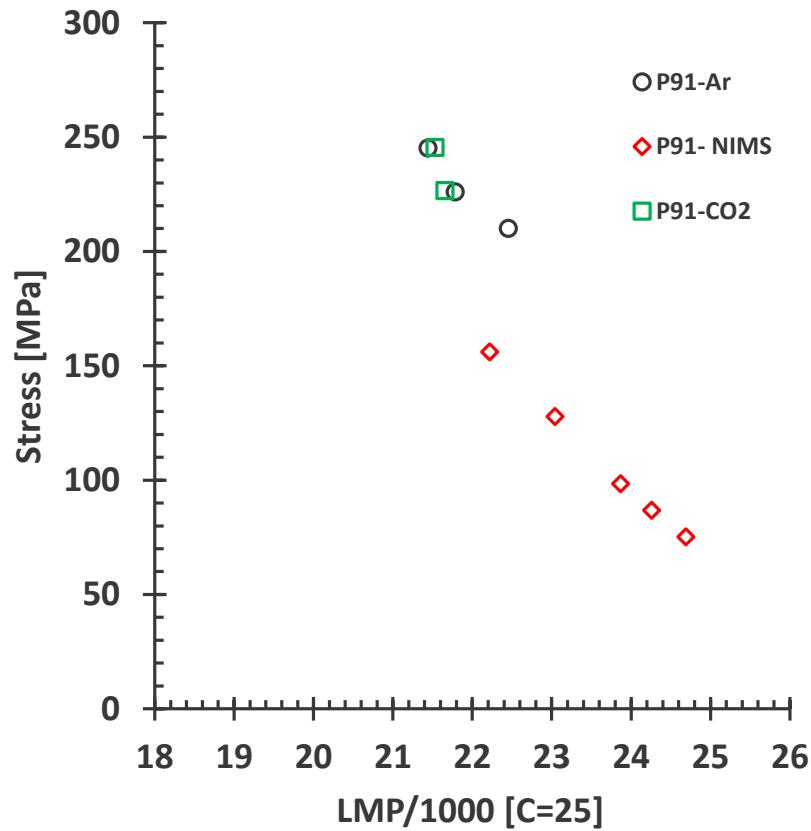
# Creep Results

- Rates calculated from cross head displacement
- Stress intervals
  - 205, 226, 245 Mpa
  - **Red-Yellow shades in 50 SCCM Ar flow**
  - **Black to Grey in 50 SCCM CO<sub>2</sub> flow**
  - Darker shades higher stress
- While some scatter in time to failure is observed minimum creep rates and rupture time is consistent
  - Lower creep rates and longer rupture times for lower stresses
- No large effect of CO<sub>2</sub> relative to Ar



# Creep Results

- $LMP = T \cdot (\log t_r + C)$
- Larson-Miller curve shows increase in performance relative to NIMS reported P91
- L-M shows no change in performance of P91 creep in  $\text{CO}_2$  relative Ar
- Monkman-Grant shows linearized behavior, with  $\text{CO}_2$  performance in line with Ar



# Accuracy

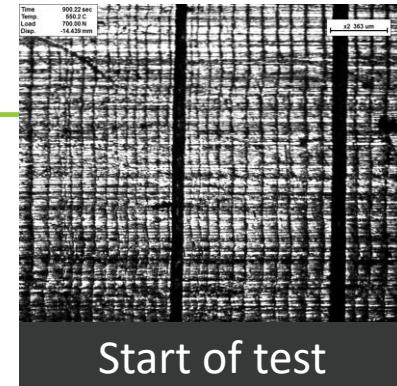
From tensile testing:

- Manual measurements
- Best practices had repeatability to within 5 pixels
- Strain uncertainty estimated at 2.5%
- Reasonable rates were measured, with a bit of uncertainty
- Difficult to measure strain with corrosion layer in CO<sub>2</sub> experiments

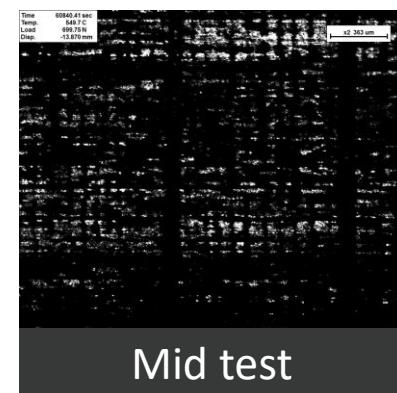
Note we encountered some software related issues.

- Images were not recoverable from a few tests
- One test had to be restarted

Magnification	Length per pixel	Focus uncertainty	Approximate gage length	Total uncertainty
-	[μm]	[μm]	[μm]	[%]
2x	2.4	12	950	2.5
25x	0.19	3	120	3.3
50x	0.094	1.5	80	2.7



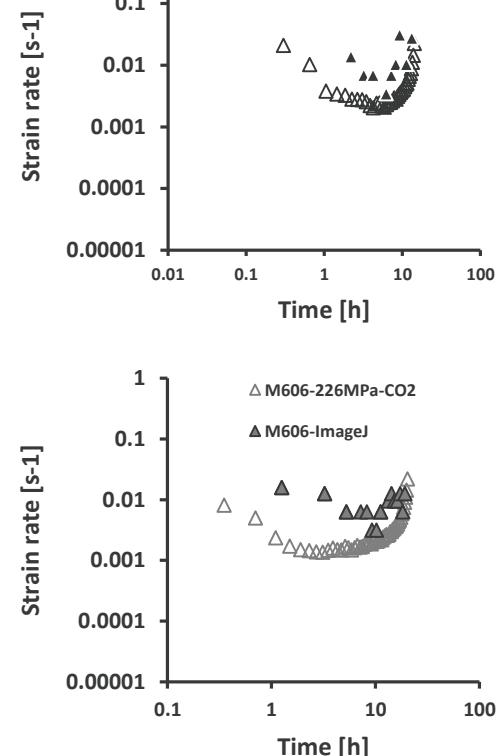
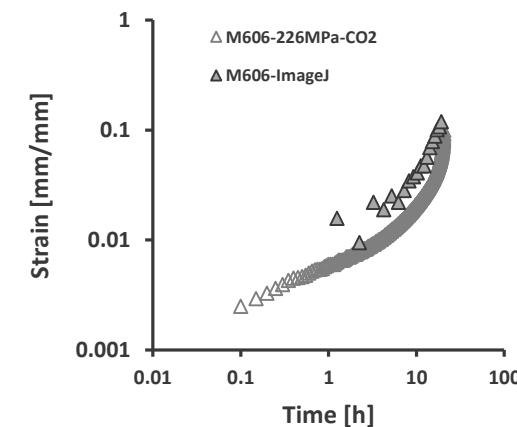
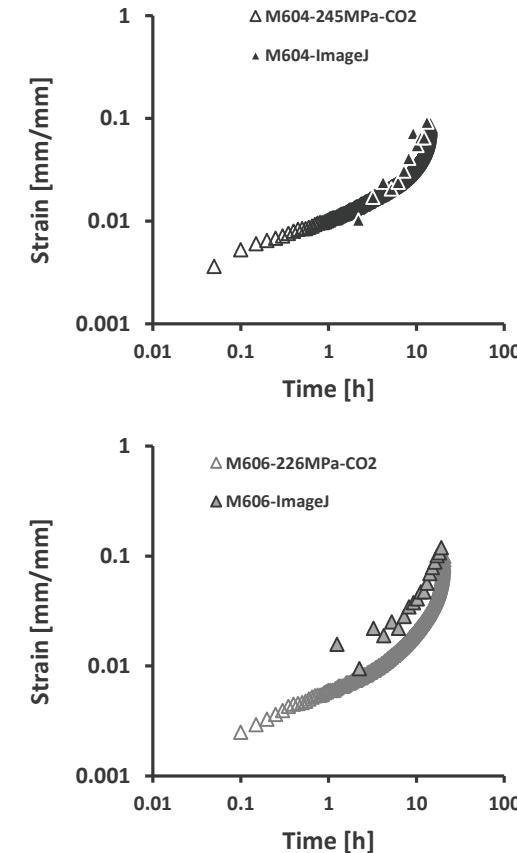
Start of test



Mid test



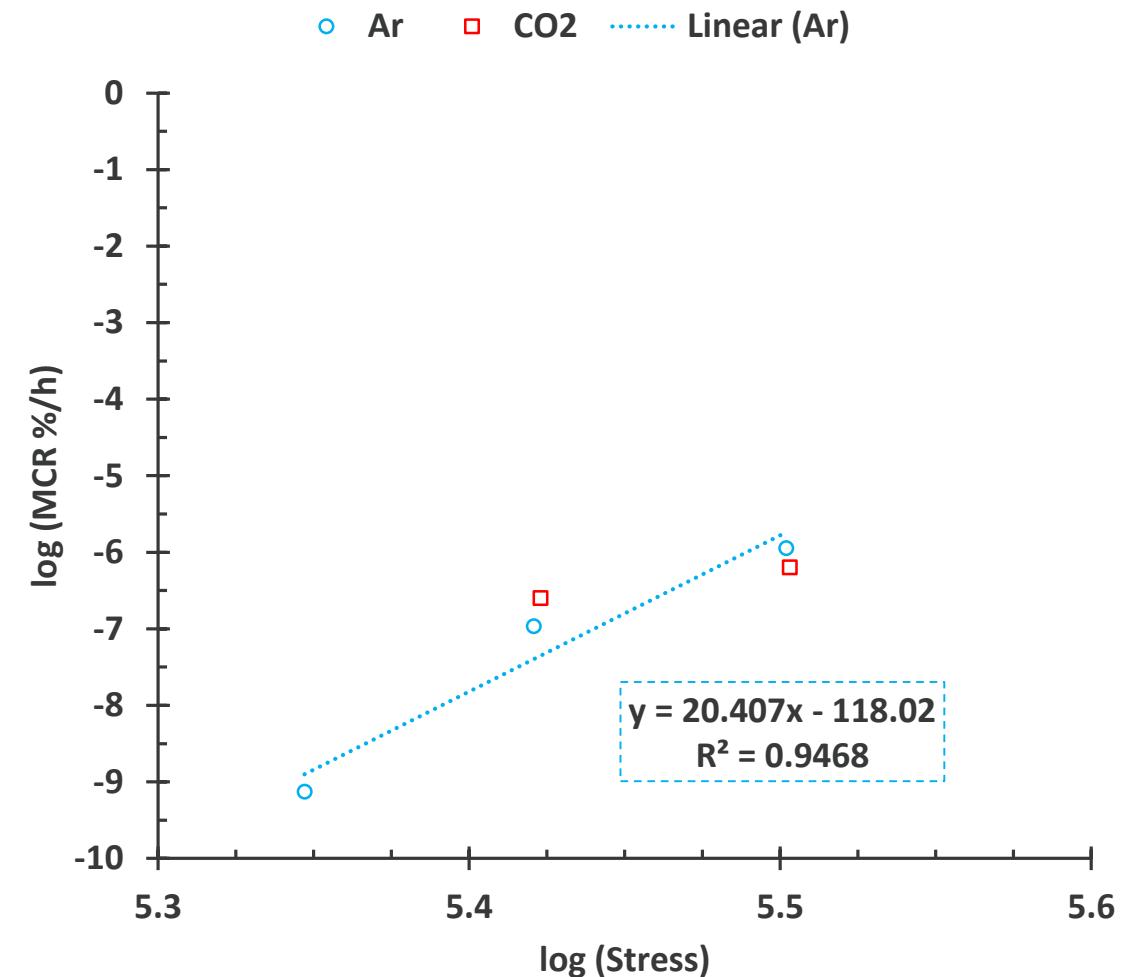
Near end test



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# Creep Exponent

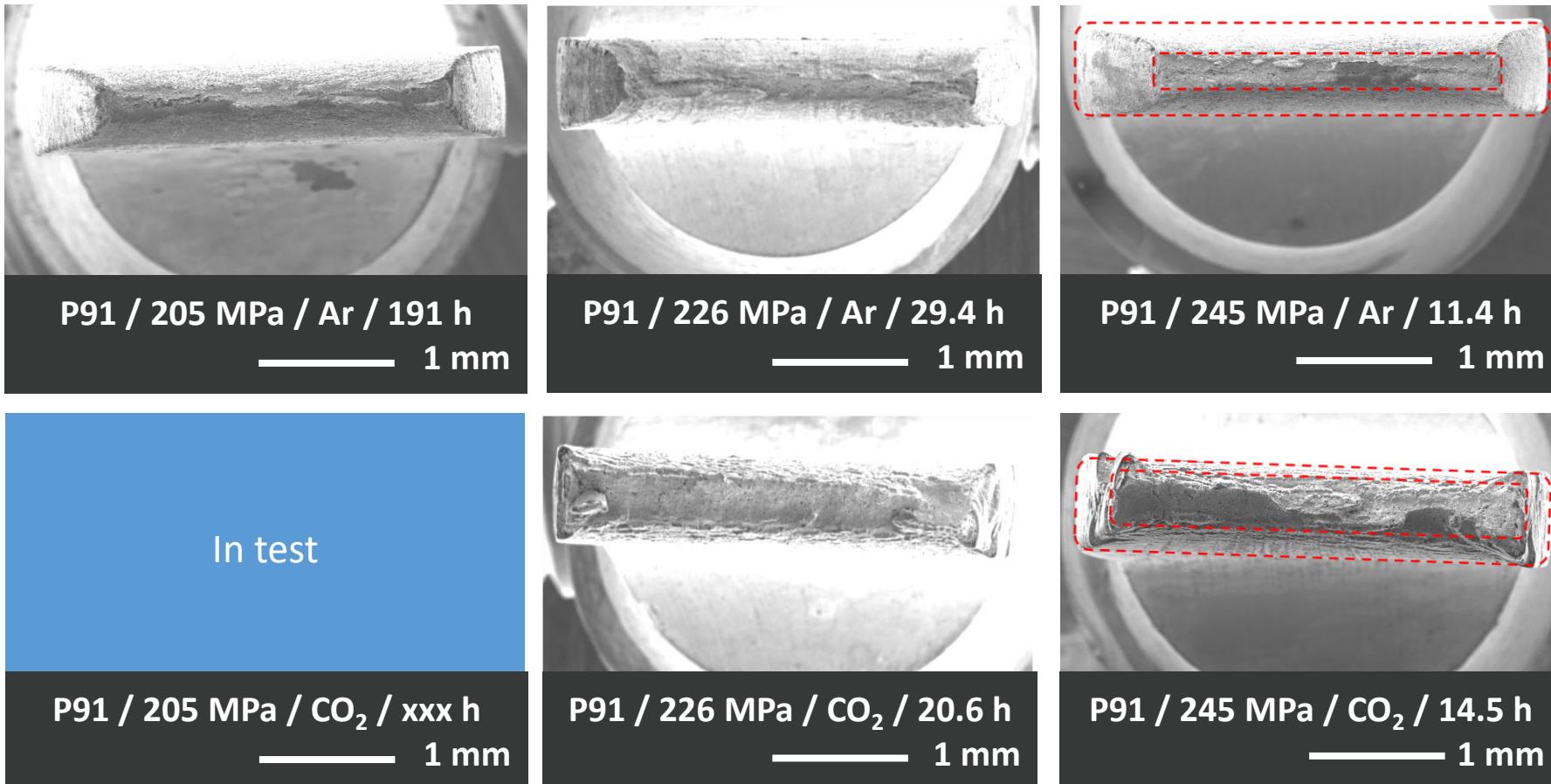
- Multiple models, Typically fit with exponential and Arrhenius fits
  - $\dot{\epsilon}_{ss} = K \cdot \sigma^n \cdot \exp(-Q/RT)$
  - $1 < n < 2$  | creep mechanism is diffusion dominated
  - $3 < n < 7$  | creep mechanism is dislocation dominated (climb)
  - $n \gg 7$  | creep mechanism approaches ideal plastic behavior (glide)
- $n \sim 20.4$  for Ar exposures
  - Waiting on creep result at 205 MPa for n-value in  $\text{CO}_2$
- High stress, short time experiments are more akin to slow strain rate tensile test than creep experiment
- Large 'n' values justified



# Fractography

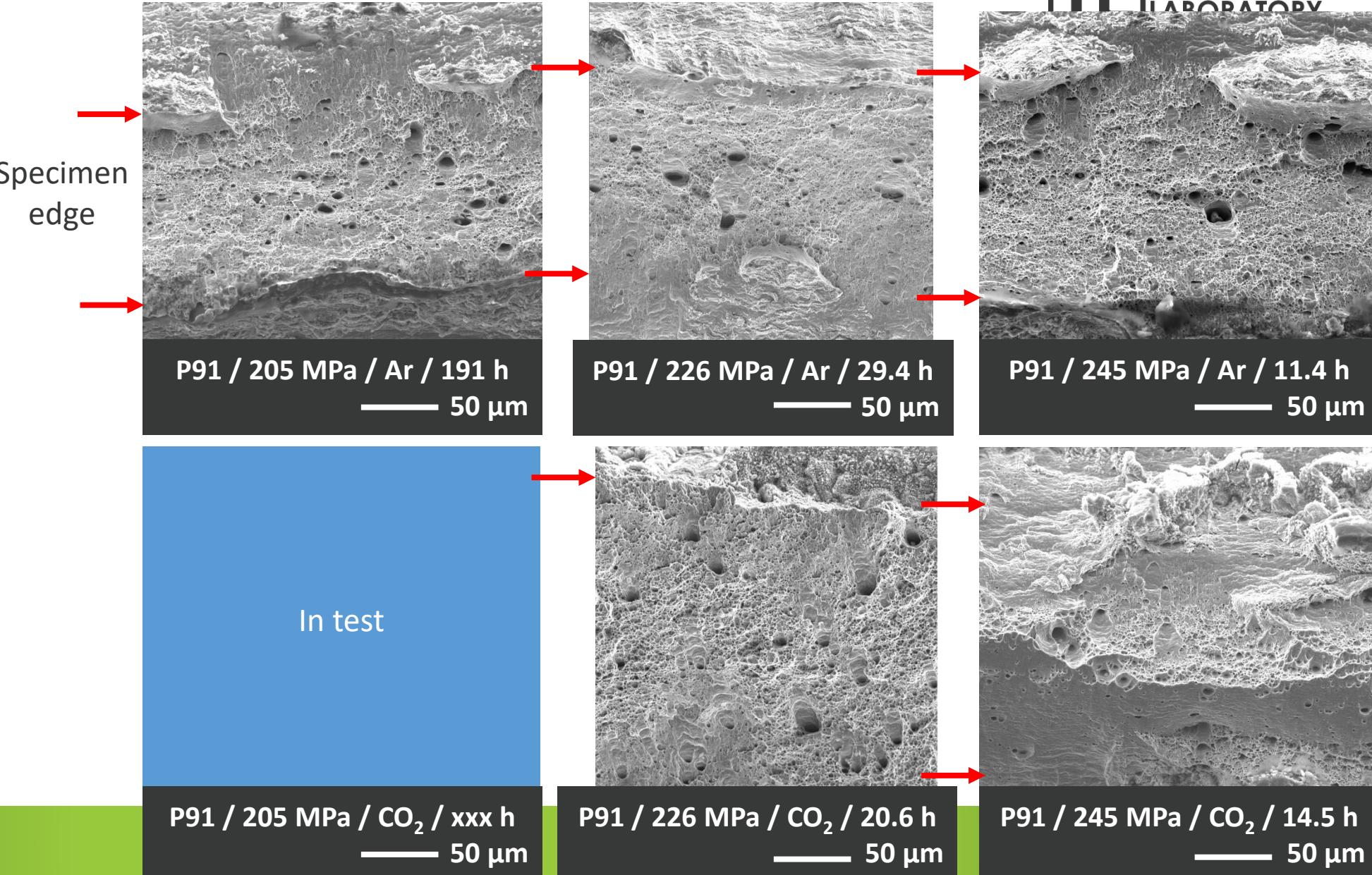
- Ductile fracture surfaces observed independent of environment
- Less reduction of area in CO<sub>2</sub>
- Highlighted in red

Env.	Stress	TTF	Elong.	RA
-	[MPa]	[h]	[%]	[%]
Ar	226.1	29.4	11%	73%
Ar	245.2	11.4	13%	66%
Ar	210	191.1	11%	79%
CO <sub>2</sub>	245.5	14.6	10%	53%
CO <sub>2</sub>	226.6	20.7	10%	50%
CO <sub>2</sub>	210	-	-	-



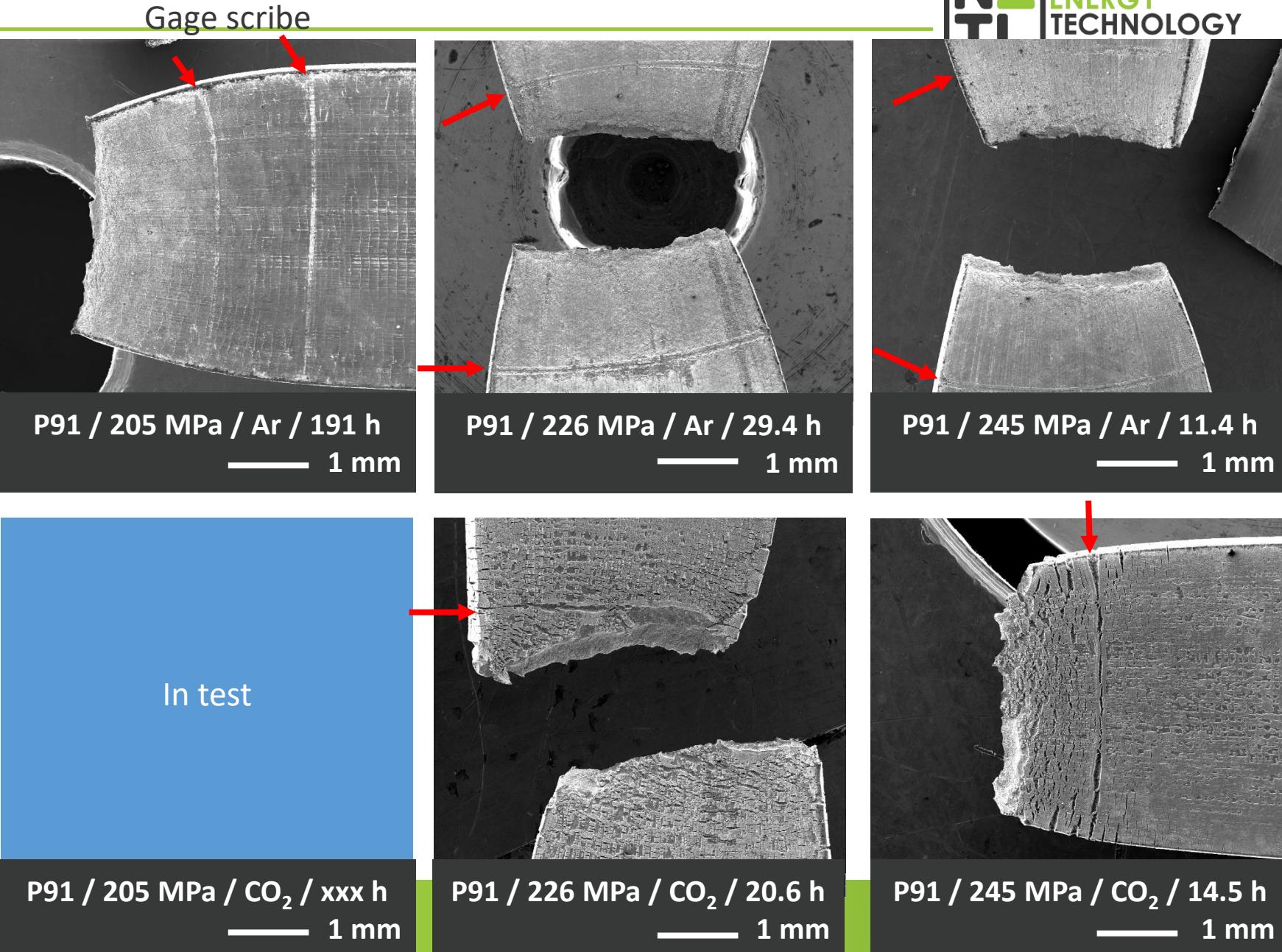
# Fractography

- Only see signs of ductile failure
- Arrows indicate specimen edge



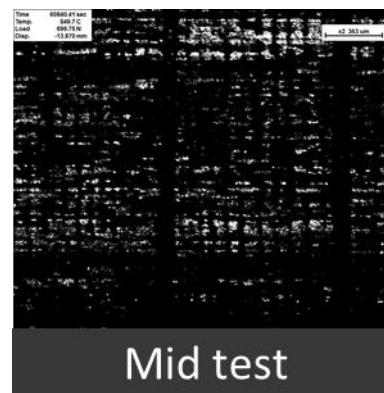
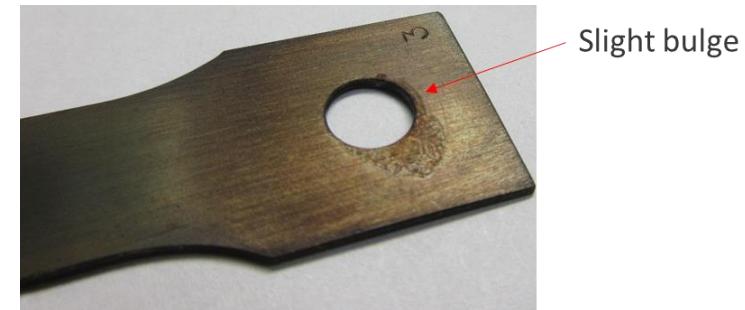
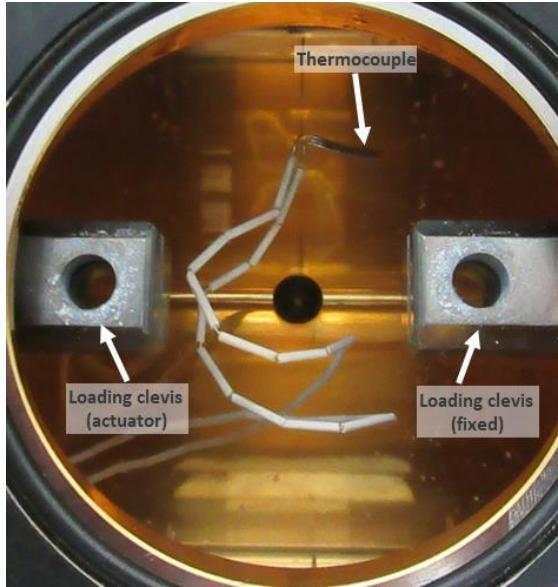
# Fractography

- Signs of corrosion cracking on  $\text{CO}_2$  specimen
- Arrows indicate gage scribe used to measure strain

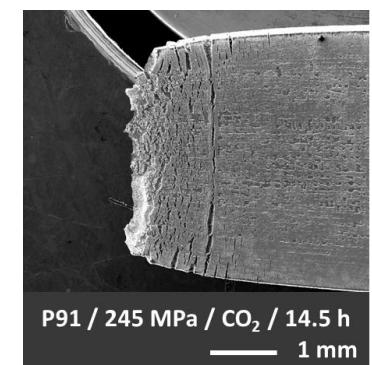
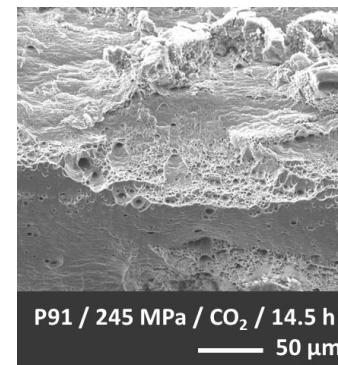
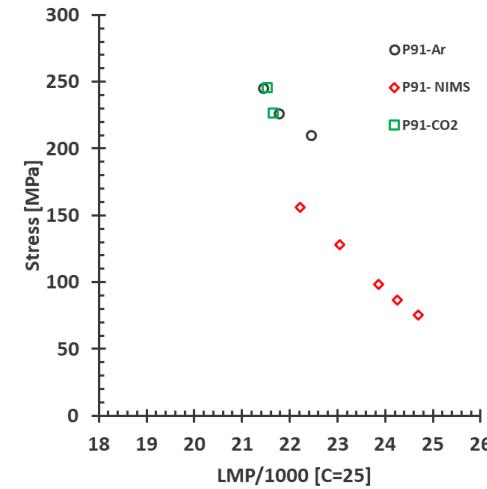


# Review

- Microscope system equipped with environmental chamber successfully tested creep performance of P91 steel
  - One must used 'compression sample' design, with threaded M6x1 ends
  - Note a few software issues (crashes/instabilities after long times)
  - More corrosive environments are unable to maintain contrast for imaging, need alternative way to place gage marks
- Automated measuring may increase accuracy of strain rates
- Balance between file size and data capture rate (GB per test)



- Tested P91 at 245, 226 and 210 MPa at 550°C in flowing Ar and CO<sub>2</sub> gas
  - No change in LMP, or minimum creep rates for CO<sub>2</sub> vs Ar flow
  - High stress exponent suggests mechanical performance analogues to a slow strain rate tensile test
  - No noted change in elongation, however, reduction of area was less for CO<sub>2</sub> tested specimens
  - Fracture surfaces show only ductile features
  - More surface cracking on specimens exposed to CO<sub>2</sub> flow
- As the LMP was higher for both CO<sub>2</sub> and Ar flow, we are looking to do creep testing in lab air to investigate the cause of better creep performance relative to NIMS data



# Thank you

Questions?



## ACKNOWLEDGEMENTS

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**Kyle Rozman, Ph.D.**

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