

Kinetic Monte Carlo Simulations of Structural Evolution of Additively Manufactured Materials

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Additive Manufacturing

Advantages

- Rapid prototyping
- Intelligent material design
- Low waste production
- Increased energy efficiency
- Reduced material cost

R&D Needs

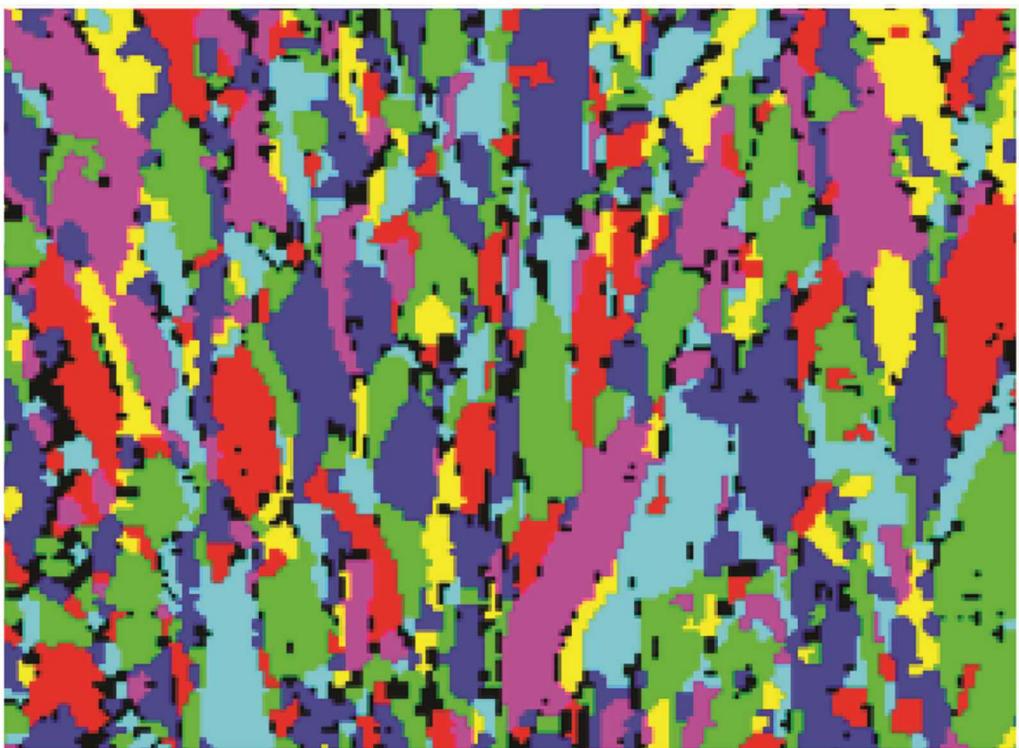
- Insufficient data
- Large uncertainty margin
- Reduced stability
- Poor understanding

Predictive models are needed to promote understanding!

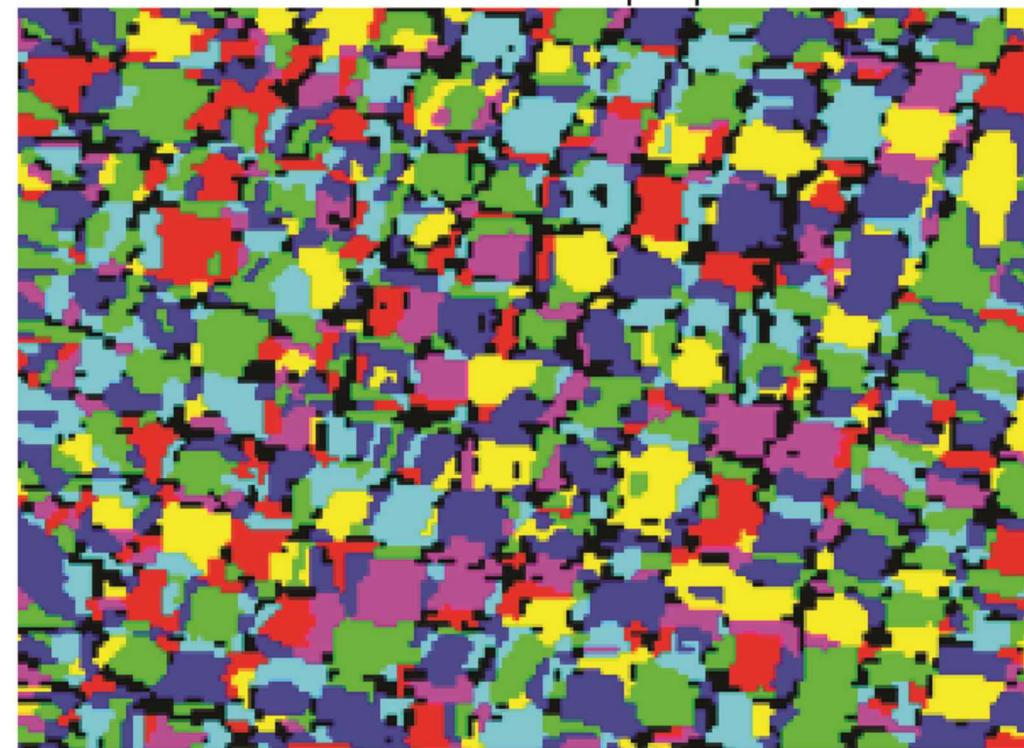


An Example of Microstructure Evolution

(a) prior to annealing



(b) after annealing



↔ molten-zone size

↔ inter-pass layer thickness

100 μm

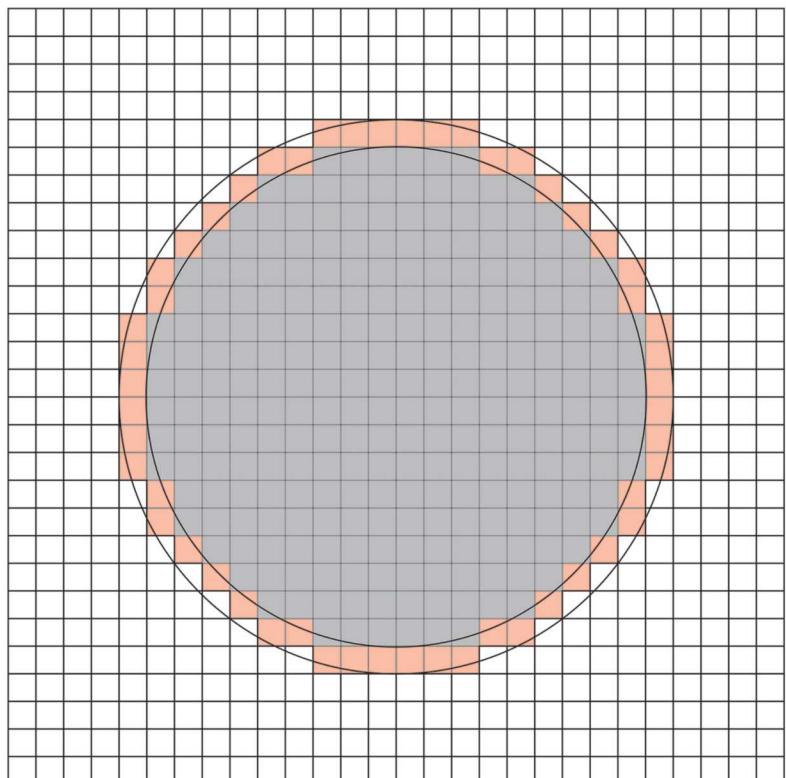
- Sample: 316L from PBF
- Annealing: from 25 to 1000°C at 6°C/min
- Analysis: EBSD

Objective: Predictive Simulations of Microstructural Evolution

- Develop a kinetic Monte Carlo (kMC) method incorporating:
 - Real time and length scales
 - Grain growth under driving energies
 - Recrystallization
- Identify model parameters
- Simulate and explain the observed microstructure evolution of the 316L PBF sample during annealing



Potts kMC Grain Growth Model



$$P = \begin{cases} \exp\left(-\frac{\Delta E_\sigma}{kT}\right), & \Delta E_\sigma > 0 \\ 1, & \Delta E_\sigma \leq 0 \end{cases}$$

- A sample is divided into sites
- Each site has an integer number representing grain orientation
- Neighboring sites with different numbers bound a grain boundary
- Flipping number causes a grain boundary energy change ΔE_σ
- Continuously flipping the spin numbers at probability P

Our Modified Grain Growth Equations

- **Grain boundary mobility**

$$m = m_0 \exp\left(-\frac{Q}{kT}\right)$$

- **Activation energy modified by grain boundary energy change**

$$\Delta Q_\sigma = c_\sigma \Delta E_\sigma$$

- **Grain orientation flipping rate (rather than probability)**

$$\Gamma_i = \begin{cases} \frac{\sigma m_0}{\Delta x^2} \exp\left(-\frac{Q + \Delta Q_\sigma}{kT}\right), & Q + \Delta Q_\sigma > 0 \\ \frac{\sigma m_0}{\Delta x^2}, & Q + \Delta Q_\sigma \leq 0 \end{cases}$$



Further Addition of Driving Energies

- Each site is associated with an additional float number representing local stored energy
- Activation energy further modified if flipping a grain number also release local stored energy $c_e \Delta E_e$

$$\Delta Q_e = c_e \Delta E_e$$

- Grain orientation flipping rate is then

$$\Gamma_i = \begin{cases} \frac{\sigma m_0}{\Delta x^2} \exp\left(-\frac{Q + \Delta Q_\sigma + \Delta Q_e}{kT}\right), & Q + \Delta Q_\sigma + \Delta Q_e > 0 \\ \frac{\sigma m_0}{\Delta x^2}, & Q + \Delta Q_\sigma + \Delta Q_e \leq 0 \end{cases}$$



kMC Simulation Steps

- Calculate flipping frequencies Γ_i^j for all sites i and all neighbors j
- Align Γ_i^j in a one-dimensional chain from 0 to $\sum_{i,j} \Gamma_i^j$. Generate a random number between 0 to $\sum_{i,j} \Gamma_i^j$. This random number points to a specific Γ_i^j event. Execute this event
- March the time forward by $dt = -\ln(R)/\sum_{i,j} \Gamma_i^j$ [1], where R is a random number between 0 and 1
- Repeat the process until total simulated time is consumed

1. A. B. Bortz, M. H. Kalos, and J. L. Lebowitz, A new algorithm for Monte Carlo simulation of Ising spin systems. J. Comput. Phys. 17, 10-18 (1975).



Recrystallization Model

□ Nucleation event: assigning a new spin number and resetting stored energy to zero

□ Nucleation rate [1]

$$\Gamma_i^n = \dot{n}_0 \exp\left(-\frac{\Delta G^*}{kT}\right)$$

□ Gibbs free energy of activation [1]

$$\Delta G^* = \frac{16\pi\sigma_n^3}{3e_i^2}$$

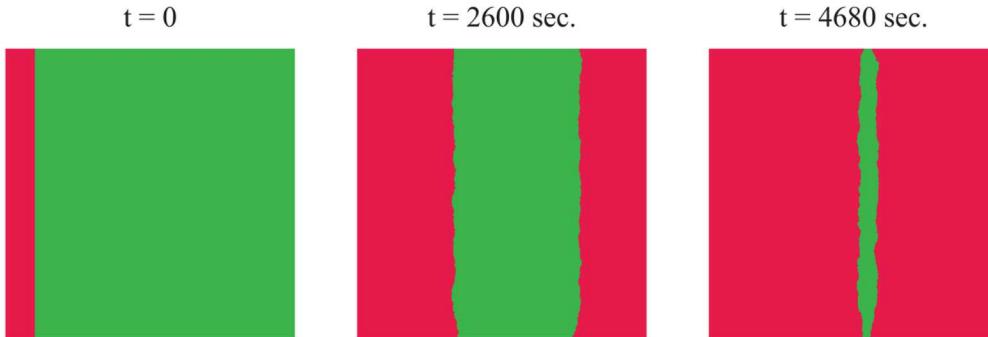
□ The same kMC steps as described above is used except that $\sum_{i,j} \Gamma_i^j$ is replaced by $\sum_{i,j} \Gamma_i^j + \sum_{i,ei \neq 0} \Gamma_i^n$

1. R. P. Sear, Nucleation: Theory and applications to protein solution and colloidal suspensions. *J. Phys.: Condens. Matter*, 19, 033101-28 (2007).

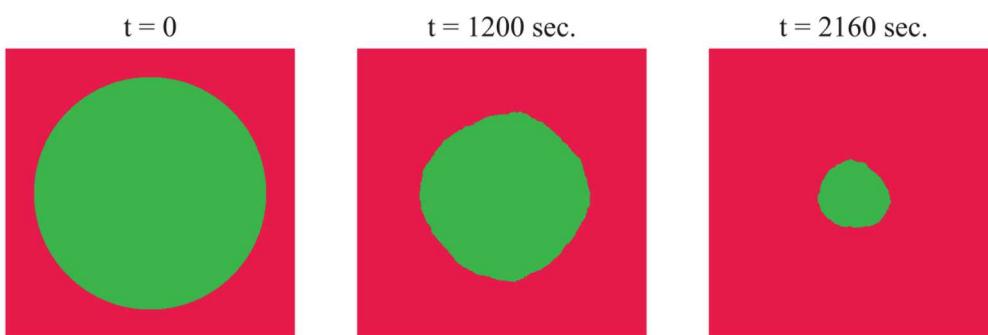


Validation Tests

(a) planar green domain at $T = 550$ K and $e_i = 0.01$ eV/Å³



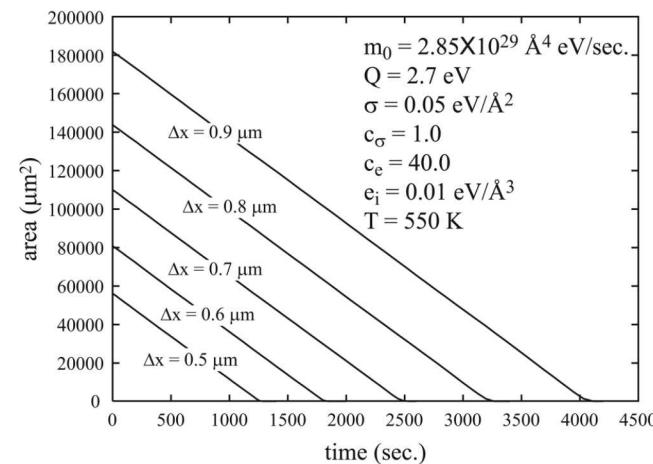
(b) circular green domain at $T = 700$ K and $e_i = 0.00$ eV/Å³



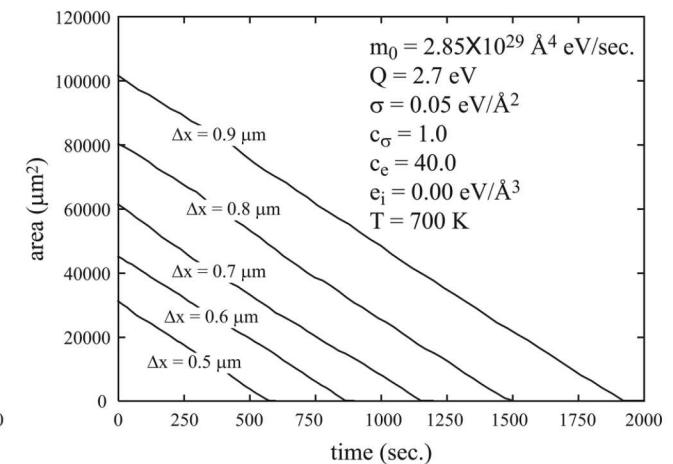
$m_0 = 2.85 \times 10^{29} \text{ \AA}^4 \text{ eV/sec.}$, $Q = 2.7 \text{ eV}$, $\sigma = 0.05 \text{ eV/\AA}^2$, $c_\sigma = 1.0$, $c_e = 40.0$ 100 \mu m

kMC simulations produced the expected results for a variety of test conditions.

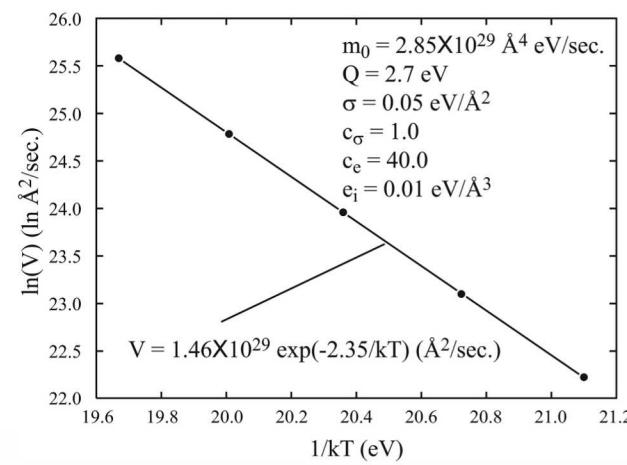
(a) planar case



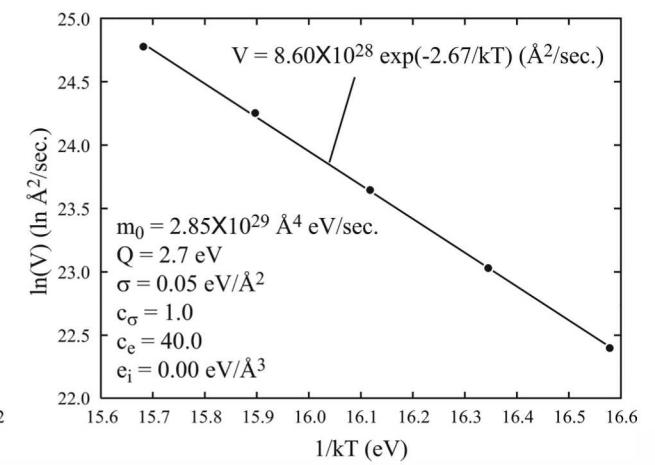
(b) circular case



(a) planar case

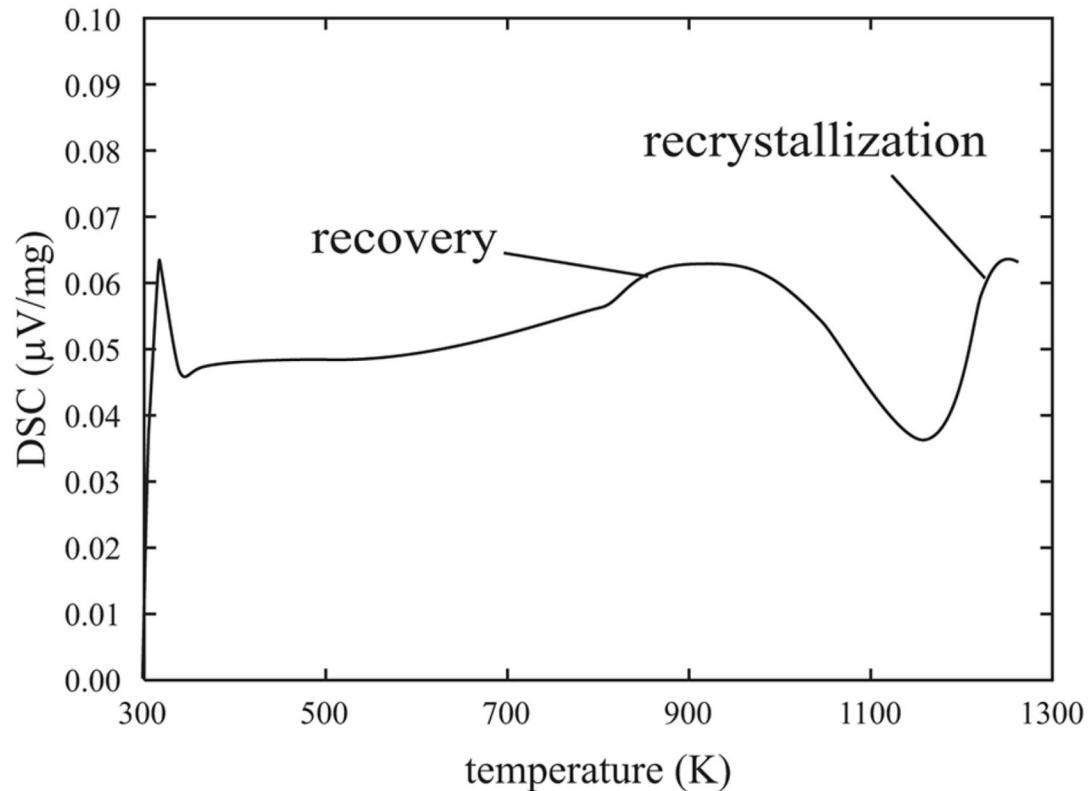


(b) circular case

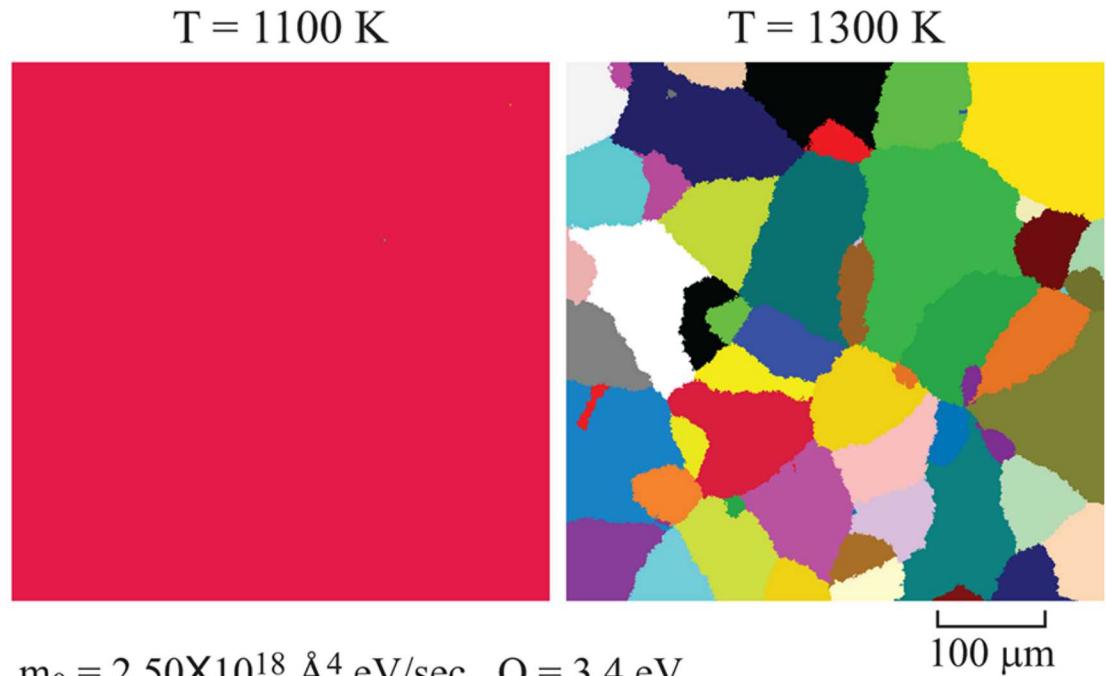


Choices of Parameters

(a) DSC experimental results



(b) kMC from 300 K to 1300 K at 6 K/min.

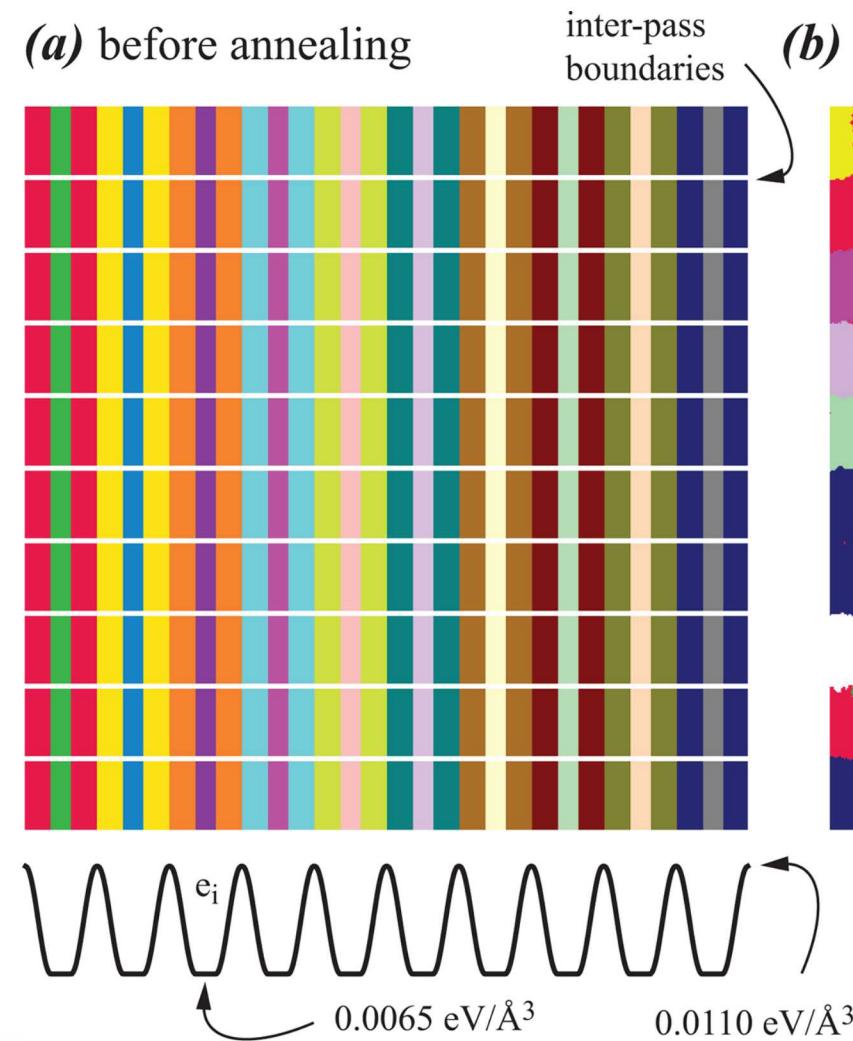


$m_0 = 2.50 \times 10^{18} \text{ \AA}^4 \text{ eV/sec.}$, $Q = 3.4 \text{ eV}$
 $\sigma = 0.05 \text{ eV/\AA}^2$, $c_\sigma = 1.0$, $c_e = 100.0$, $\dot{n}_0 = 1.00 \times 10^{13} / \text{sec.}$
 $\sigma_n = 0.032 \text{ eV/\AA}^2$, $e_i = 0.011 \text{ eV/\AA}^3$

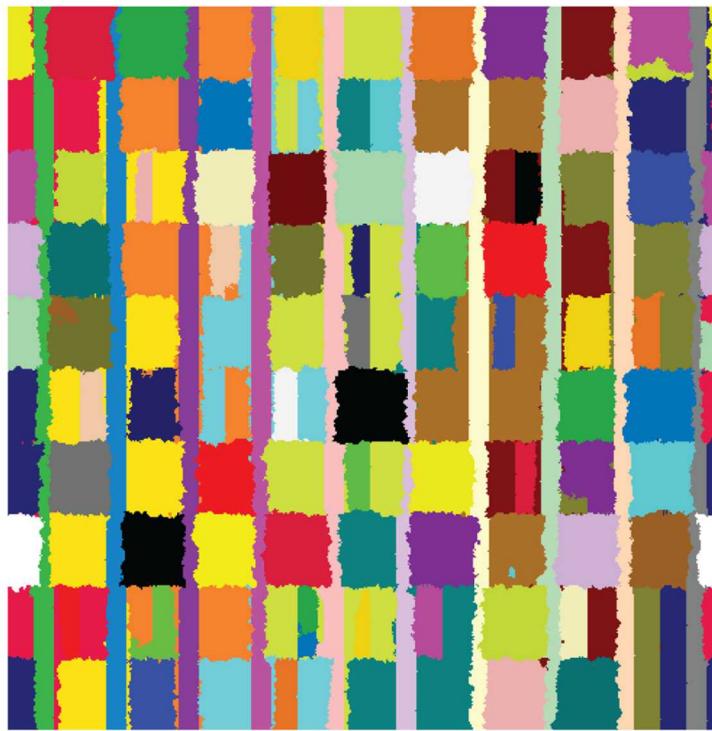
kMC simulation of a 316L sample annealed from 300 K to 1300 K at 6 K/min.

Predicted Microstructure Evolution

(a) before annealing



(b) after annealing at 6 K/min. to 1300 K



$$\begin{aligned}m_0 &= 2.50 \times 10^{18} \text{ Å}^4 \text{ eV/sec.}, Q = 3.4 \text{ eV} \\ \sigma &= 0.05 \text{ eV/Å}^2, c_\sigma = 1.0, c_e = 100.0 \\ \dot{n}_0 &= 1.00 \times 10^{13} / \text{sec.}, \sigma_n = 0.032 \text{ eV/Å}^2\end{aligned}$$

- The PBF produced 316L has stored energy periodically
- Model parameters are chosen to give phenomena seen from the DSC experiments
- Simulations predict the formation of cubic grain

Summaries

- A new kMC method with real time and length scales has been developed
- This kMC method incorporates grain growth, grain growth under a distribution of stored energy, and recrystallization
- The kMC parameters can be determined from DSC experiments during annealing from 300 to 1300 K
- Our kMC simulations indicate that periodically stored energy in a PBF manufactured 316L sample can cause the formation of cubic grains during annealing

