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Mechanochemical Nonhydrolytic Sol–Gel-Strategy for the Production of Mesoporous Multimetallic Oxides

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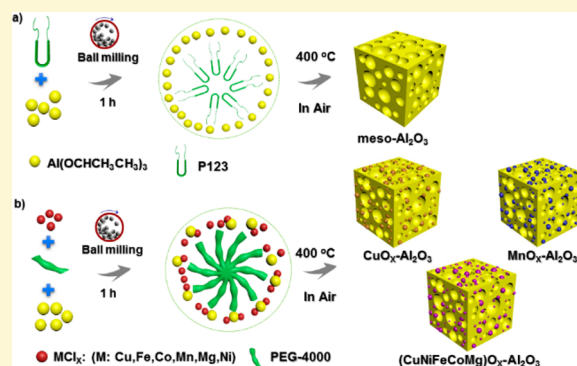
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Supporting Information

ABSTRACT: Mesoporous metal oxides with wide pore size, high surface area, and uniform porous structures have demonstrated excellent advantages in various fields. However, the state-of-art synthesis approaches are dominated by wet chemistry, accompanied by use of excessive solvent, and the requirement of time-consuming drying process. Herein, we report a mechanochemical solid-state route to synthesize mesoporous Al_2O_3 (meso- Al_2O_3) via aluminum isopropoxide-copolymers assembly. The obtained meso- Al_2O_3 shows a record high surface area ($\sim 644 \text{ m}^2 \text{ g}^{-1}$) and narrow pore size distribution (centered at $\sim 5 \text{ nm}$). Moreover, a mechanochemical nonhydrolytic sol–gel strategy is introduced to fabricate mesoporous transition metal (Cu, Co, Mn, Fe, Mg, Ni)-aluminum binary oxide by using anhydrous metal chlorides and aluminum isopropoxide interplay. More importantly, four or five metals-aluminum oxide complexes with abundant mesopores and single cubic crystalline phase known as high-entropy ceramics are produced. To the best of our knowledge, mesoporous high-entropy metal oxides have not been prepared before, because the high crystallization temperature would make mesopores collapse. Additionally, this high-entropy property endows $(\text{CuNiFeCoMg})\text{O}_x\text{-Al}_2\text{O}_3$ with superior SO_2 -resisting performance (1000 ppm of SO_2 in N_2 at 280°C) in the catalytic oxidation of CO compared to single $\text{CuO-Al}_2\text{O}_3$.



1. INTRODUCTION

Mesoporous metal oxides with wide pore size, high surface area, uniform pore size distribution, as well as a variety of structures and compositions endow them particularly attractive for applications in catalysis, adsorption, battery technology, sensing, and so on.^{1–6} Template-assisted processes via either soft- or hard templates were usually used for the synthesis of mesoporous metal oxides.^{7–12} In general, the hard-templating technique relied on mesoporous silica or carbon aerogel as the hard templates, which inevitably required additional steps and costs.^{9,13–21} In comparison, organic–inorganic assembly routes, using block copolymers or surfactants as soft templates have been developed as a more straightforward method.^{22–26} It is undeniable that the solvent evaporation-induced self-assembly method has been successfully applied into the synthesis of many

mesoporous metal oxides.^{27–29} However, several issues seem to remain in this wet templating process, including: (1) excessive organic solvents are used; (2) it requires the metal precursors that have to be dissolved in the solvents, which limits the applications of many insoluble metal precursors;^{3,30} and (3) the slow solvent evaporation is a time-consuming step.³¹

To resolve these disadvantages, solid-state synthesis, an old method in material processing, has been revisited.^{32–35} Several porous materials, such as zeolite, metal–organic frameworks, covalent–organic frameworks, ordered mesoporous polymers, and mesoporous carbons have been already synthesized via

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59 mechanochemical processes.^{36–38} Although a solvent-free as-
60 sembly method has also been introduced for the construction of
61 mesoporous metal oxide (TiO_2), this route needs HCl as the
62 solvent and aged at 140 °C for 24 h before calcination, making it
63 not a “true” solvent-free system.³⁷ As a result, the synthesis of
64 mesoporous metal oxides via mechanochemical solid-state
65 method, especially for metal oxides hybrids, is highly welcome.
66 Moreover, alumina has been widely used as an important carrier
67 in catalysis, and alumina-based mixed mesoporous metal oxides
68 also display extensive application prospect.^{39–43} Therefore, the
69 one-pot preparation of mesoporous alumina-supported metal
70 oxides via mechanochemical solid-state method seems like a
71 more efficient and straightforward pathway toward industrial
72 catalysis.

73 Herein, we demonstrate the facile, rapid, and solid-state
74 synthesis of mesoporous aluminum oxide (meso- Al_2O_3) by a
75 mechanochemical assembly between aluminum isopropoxide
76 and commercial polymers (e.g., PEO-PPO-PEO, P123; poly-
77 ethylene glycol, PEO, PEG). Moreover, binary Al-based oxides
78 (Cu, Co, Mn et al.) with abundant mesopores can be easily
79 constructed by mechanochemical nonhydrolytic sol–gel proc-
80 ess (NHSG, metal chlorides and aluminum isopropoxide
81 interplay).^{44–46} More interestingly, four or five transition
82 metal species-aluminum oxide hybrids—the so-called high-
83 entropy oxides—with single cubic crystalline phase and rich
84 mesoporous structure were synthesized via this mechanochem-
85 ical NHSG method. To our surprise, the high-entropy feature
86 endows $(\text{CuNiFeCoMg})_x\text{-Al}_2\text{O}_3$ with superior SO_2 -resisting
87 performance (1000 ppm of SO_2 in N_2 at 280 °C) in the catalytic
88 oxidation of CO compared to $\text{CuO-Al}_2\text{O}_3$.

2. EXPERIMENTAL SECTION

89 **2.1. Catalyst Synthesis.** **2.1.1. meso- Al_2O_3 .** One gram of metal
90 precursor (aluminum isopropoxide) and 0.2–0.4 g soft templates
91 (Pluronic P123: 5800 g mol^{-1} ; PEG: 4000 g mol^{-1} , Pluronic F127
92 (F127), and cetyltrimethylammonium bromide (CTAB)) were added
93 in a 9.5 mL screw-capped iron reactor with two big (diameter: 1.2 cm)
94 and three small (diameter: 0.5 cm) iron steel ball bearings. The sealed
95 reactor was placed in a high-speed vibrating ball miller (Retsch
96 MM400), and the mixtures were ball milled for 60 min at a vibrational
97 frequency of 30 Hz. The resulting samples were washed using ethanol
98 and dried at 40 °C under vacuum conditions, followed by calcination at
99 400 °C with a heating rate of 2 °C min^{-1} .

100 **2.1.2. Mesoporous Binary Metal Oxides (Cu, Co, and Mn et al.).**
101 Aluminum isopropoxide (2.67 mmol) and a transition metal chloride
102 (4 mmol) as well as 0.4 g PEG were added in a 9.5 mL screw-capped
103 iron reactor with two big (diameter: 1.2 cm) and three small (diameter:
104 0.5 cm) iron steel ball bearings. After ball milling for 60 min, the post-
105 treatment procedure is the same as that with the synthesis of meso-
106 Al_2O_3 . All catalysts were washed using water for three times before use.

107 **2.1.3. $(\text{CuNiFeCo})_x\text{-Al}_2\text{O}_3$.** Aluminum isopropoxide (2.67 mmol),
108 copper chloride (1 mmol), nickel chloride (1 mmol), cobalt chloride (1
109 mmol), and ferrous chloride (1 mmol), as well as 0.4 g PEG are added
110 in a 9.5 mL screw-capped iron reactor with two big (diameter: 1.2
111 cm) and three small (diameter: 0.5 cm) iron steel ball bearings. After
112 ball milling for 60 min, the post-treatment procedure is the same as that
113 with the synthesis of meso- Al_2O_3 .

114 **2.1.4. $(\text{CuNiFeCoMg})_x\text{-Al}_2\text{O}_3$.** Aluminum isopropoxide (3.33
115 mmol), copper chloride (1 mmol), nickel chloride (1 mmol), cobalt
116 chloride (1 mmol), ferrous chloride (1 mmol), and magnesium chloride
117 (1 mmol), as well as 0.4 g PEG are added in a 9.5 mL screw-capped
118 iron reactor with two big (diameter: 1.2 cm) and three small (diameter:
119 0.5 cm) iron steel ball bearings. After ball milling for 60 min, the post-
120 treatment procedure is same with the synthesis of meso- Al_2O_3 .

121 **2.2. Characterizations.** X-ray diffraction (XRD) was collected on a
122 PANalytical Empyrean diffractometer operated at 45 kV and 40 mA

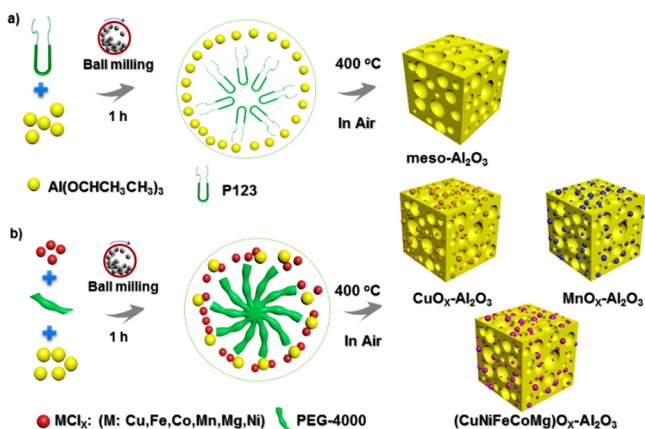
(scanning step: 0.02° per step). N_2 adsorption–desorption isotherms 123
were analyzed on a TriStar 3000 volumetric adsorption analyzer 124
manufactured and by Micromeritics Instrument Corp. A classical 125
Barrett–Joyner–Halenda (BJH) method was employed to determine 126
the pore size distribution by using adsorption branch of isotherms. Prior 127
to analysis, all samples were activated at 160 °C for 8 h. Scanning 128
electron microscopy (SEM) with energy-dispersive X-ray spectroscopy 129
(EDS) mapping results were recorded using a Zeiss EVO-MA15 130
scanning electron microscope with a Bruker XFlash 6130 detector. 131
Scanning electron microscopy (SEM) images were recorded using a 132
Zeiss AURIGA. Scanning transmission electron microscopy (STEM) 133
was performed on a Nion microscope operated at 200 kV and ARM 134
200CF equipped with a probe corrector. The images are recorded using 135
high angle annular dark field (HAADF) and bright field (BF) imaging. 136
X-ray photoelectron spectroscopy (XPS) was performed by AXIS Ultra 137
DLD spectrometer (Kratos, Japan). The obtained data were calibrated 138
by C 1s standard peak and analyzed by Casa XPS software. 139
Temperature-programmed desorption (TPD) of CO was recorded 140
using FineSorb-3010 from Zhejiang Finetec Instruments Co., Ltd. with 141
a thermal conductivity detector (TCD). Prior to characterization, fresh 142
and spent catalysts were preheated at 120 °C under argon atmosphere. 143
Elemental analysis of the samples was done by inductively coupled- 144
plasma atomic emission spectroscopy (ICP-AES) using the Optima 145
2100 DV spectrometer (PerkinElmer Corporation). The actual metal 146
ratios are summarized in Table S1 of the Supporting Information (SI). 147
Fourier Transform infrared spectroscopy (FTIR) spectra were 148
collected on a Bio-Rad FTIR spectrometer (Excalibur series). 149

2.3. Experimental Procedures. CO oxidation experiments were 150
carried out in a fixed bed reactor (straight quartz tube with 4 mm inner 151
diameter) at atmospheric pressure. For the measurement of CO light- 152
off curves showing CO conversion as a function of reaction 153
temperature, 20 mg of catalyst supported by quartz wool was loaded 154
in the reactor. The feed gas of 1% CO balanced with dry air (<4 ppm 155
water) passed through the catalyst bed at a flow rate of 10 mL/min 156
corresponding to a gas hourly space velocity (GHSV) of 30 000 mL/(h 157
g cat). The concentrations of CO and CO_2 in the reactor were analyzed 158
by a Buck Scientific 910 gas chromatograph equipped with a dual 159
molecular sieve/porous polymer column (Alltech CTR1) and a thermal 160
conductivity detector. CO conversion is calculated as the peak areas of 161
CO of consumption divided by that of CO before the reaction. 162

3. RESULTS AND DISCUSSION

Scheme 1 summarizes a mechanochemical route for the 163 s1
synthesis of meso- Al_2O_3 and related metal oxide hybrids using 164
aluminum isopropoxide and anhydrous metal chlorides as the 165
precursors, and P123 or PEG as the soft template. An efficient 166
assembly between aluminum isopropoxide and P123 micelles is 167
realized by mechanically grinding the mixture in a vibrating ball 168

Scheme 1. Synthesis of meso- Al_2O_3 , Mesoporous Mixed Metal Oxides Using the Mechanochemical NHSG Method



169 miller for 60 min (much shorter than the traditional solvent
170 evaporation induced assembly process, 1–3 days in most
171 cases).^{23,47} It has been reported that the condensation between
172 chloride and alkoxide by mixing the chloride precursors (MCl_x)
173 and oxygen donor ($\text{Al}(\text{OR})_3$) to produce the xerogels is
174 considered to be an NHSG process.⁴⁸ The resulting xerogels are
175 then washed and dried under vacuum and finally calcined in air
176 to achieve the corresponding metal oxides. In this work, the
177 mechanochemical NHSG process between metal chlorides and
178 aluminum isopropoxide in the solid-state via complex
179 mechanism is restricted to the periphery of micelles. The
180 obtained xerogels are performed by XRD (Figure S1) and then
181 calcinated at 400 °C in air to remove template. By combining
182 with aluminum isopropoxide-copolymers cross-linking and
183 NHSG process in one-step, the mixed oxides with uniform
184 mesopores can be prepared.

185 The textural properties of the as-synthesized Al_2O_3 without
186 and with different surfactants are summarized (Table 1) by N_2

Table 1. Summary of Pore Parameter for Different Metal Oxides Calculated Using N_2 Adsorption–Desorption Isotherms at 77 K

samples	S_{BET} ($\text{m}^2 \text{g}^{-1}$)	pore volume ($\text{cm}^3 \text{g}^{-1}$)	average pore size (nm)
Al_2O_3	246	0.32	
meso- Al_2O_3 (0.4 g P123)	644	0.75	4.2
meso- Al_2O_3 (0.2 g PEG)	257	0.16	2.3
meso- Al_2O_3 (0.2 g CTAB)	380	0.26	
meso- Al_2O_3 (0.2 g F127)	329	0.24	
meso- Al_2O_3 (0.2 g P123)	415	0.34	2.6
$\text{CuO}-\text{Al}_2\text{O}_3$	115	0.11	2.8
$\text{Mn}_3\text{O}_4-\text{Al}_2\text{O}_3$	143	0.27	4.9
$\text{Co}_3\text{O}_4-\text{Al}_2\text{O}_3$	133	0.14	7.2
$\text{Fe}_2\text{O}_3-\text{Al}_2\text{O}_3$	178	0.24	5.8
$\text{NiO}-\text{Al}_2\text{O}_3$	168	0.15	3.3
$\text{MgO}-\text{Al}_2\text{O}_3$	176	0.18	6.3
$(\text{CuNiFeCo})\text{O}_x-\text{Al}_2\text{O}_3$	129	0.13	3.9
$(\text{CuNiFeCoMg})\text{O}_x-\text{Al}_2\text{O}_3$	198	0.24	4.8

^aSpecific surface area calculated using the BET equation. ^bSingle point adsorption total pore volume of pores. ^cAverage pore diameter from the distribution maxima by resolving adsorption branch with BJH model.

187 adsorption–desorption isotherms at 77 K in Figure S2. The
188 blank Al_2O_3 sample without templates offers a low surface area
189 ($246 \text{ m}^2 \text{g}^{-1}$) with a broad pore size distribution in Figure S2a,c.
190 In comparison, a meso- Al_2O_3 sample with 0.2 g P123 as the
191 template gives a type-IV isotherm with a H3-type hysteresis
192 loop, resulting in both high surface area ($415 \text{ m}^2 \text{g}^{-1}$) and
193 narrow pore size distribution centered at $\sim 3 \text{ nm}$ (Figure S2).
194 For meso- Al_2O_3 , a higher calcination temperature at 800 °C will
195 lead to the collapse of a portion of the mesopores, as seen in
196 Figure S3. Additionally, PEG, F127, and CTAB as templates also
197 contribute to the formation of mesopores smaller than 10 nm
198 (Table 1). The weight ratio of P123/Al precursor exhibits
199 significant influence for the formation of mesopores. The N_2
200 adsorption–desorption isotherms and pore size distribution
201 results of meso- Al_2O_3 synthesized with 0.4 g P123 displays a
202 higher surface area of $644 \text{ m}^2 \text{g}^{-1}$ and narrower pore size
203 distribution of $\sim 5 \text{ nm}$ (Figure 1a,b). It is noteworthy that the
204 difference between the average pore size from Table 1 and the
205 pore size distribution from Figure 1 should be ascribed to the

limitation and accuracy of BJH model used. The transmission
electron microscopy (TEM) and scanning transmission electron
microscopy (STEM) image of meso- Al_2O_3 with 0.4 g P123 as
the template also reveal the existence of a sponge-like
nanoarchitecture with a large number of apparent mesopores
(Figure 2a,b). A portion of mesopores is marked by yellow
circles from STEM-EDS image in Figure S4. The wide-angle X-
ray diffraction (XRD) result of meso- Al_2O_3 calcined at 400 °C
with different templates shows an amorphous structure (Figures
3a and S5).

Compared with monometallic oxide, binary metal oxides tend
to possess wider applications in catalysis. The traditional NHSG
method for the production of metal oxides often takes place in
organic solvents, and the corresponding pore size distribution is
much wider.^{46,48} By the mechanochemical NHSG route in the
solid-state, mesoporous binary metal oxides ($\text{CuO}-\text{Al}_2\text{O}_3$,
 $\text{Mn}_3\text{O}_4-\text{Al}_2\text{O}_3$ and $\text{Co}_3\text{O}_4-\text{Al}_2\text{O}_3$) with uniform pore size
distribution could be successfully synthesized with optimized
PEG as the optimized template (Figures 1c–h and S6). In
comparison to P123 and F127, PEG is a safe, inexpensive and
renewable material. However, PEG has been used as soft
templates with only limited success in directing mesopores until
now. XRD results reveal the clear crystalline structures of CuO
[JCPDS 44–0706], Mn_3O_4 [JCPDS 18–0803], and Co_3O_4
[JCPDS 43–1003] in $\text{CuO}-\text{Al}_2\text{O}_3$, $\text{Mn}_3\text{O}_4-\text{Al}_2\text{O}_3$, and
 $\text{Co}_3\text{O}_4-\text{Al}_2\text{O}_3$ samples, respectively (Figure 3a), and the
corresponding crystalline sizes calculated by the Scherrer
equation are 15.9, 13.4, and 11.8 nm, respectively. Additionally,
no diffraction peaks associated with Al_2O_3 can be found in three
binary metal oxides. Moreover, the diffraction peaks corre-
sponding to NiO, MgO, and Fe_2O_3 crystalline phase,
respectively, were discovered in the $\text{NiO}-\text{Al}_2\text{O}_3$, $\text{MgO}-$
 Al_2O_3 , and $\text{Fe}_2\text{O}_3-\text{Al}_2\text{O}_3$ samples, respectively, as shown in
XRD results (Figure S7).

The surface area and pore structure of $\text{CuO}-\text{Al}_2\text{O}_3$, Mn_3O_4-
 Al_2O_3 , and $\text{Co}_3\text{O}_4-\text{Al}_2\text{O}_3$ were examined with N_2 adsorption–
desorption isotherm measurements (Figure 1c,e,g). All samples
show representative type-IV curves with clear capillary
condensation steps between the relative pressure (P/P_0) of
0.4–0.8, underscoring the presence of rich mesopores. The
specific surface areas of $\text{CuO}-\text{Al}_2\text{O}_3$, $\text{Mn}_3\text{O}_4-\text{Al}_2\text{O}_3$, and
 $\text{Co}_3\text{O}_4-\text{Al}_2\text{O}_3$ are 115, 144, and $133 \text{ m}^2 \text{g}^{-1}$, respectively
(Table 1). More importantly, the pore size distributions by
resolving the adsorption branch of N_2 isotherms with Barrett–
Joyner–Halenda (BJH) method centered at 4, 6, and 7 nm for
 $\text{CuO}-\text{Al}_2\text{O}_3$, $\text{Mn}_3\text{O}_4-\text{Al}_2\text{O}_3$, and $\text{Co}_3\text{O}_4-\text{Al}_2\text{O}_3$, respectively
(Figure 1d,f,h).⁴⁹ Moreover, N_2 adsorption and pore size
distribution results of $\text{NiO}-\text{Al}_2\text{O}_3$, $\text{MgO}-\text{Al}_2\text{O}_3$, and Fe_2O_3-
 Al_2O_3 samples also reveal the existence of abundant mesoporous
structure (Table 1; Figure S8). These obtained abundant and
uniform mesopores in binary metal oxides should be ascribed to
solid-state self-assembly behavior of PEG micelles and different
metal ions, confining NHSG process within the periphery of
micelles. The results indicate that the mechanochemical NHSG
route has the traits of universality for the preparation of a variety
of binary metal oxides with uniform mesopore distributions.
Additionally, the obtained surface area for both meso- Al_2O_3 and
binary metal oxides are comparable to the previously reported
mesoporous alumina and metal containing mesoporous alumina
synthesized by conventional wet chemistry.^{24,50}

$\text{Mn}_3\text{O}_4-\text{Al}_2\text{O}_3$ is then chosen as a representative to
investigate the pore structure of binary metal oxides in a more
tangible manner. Wormhole architectures with rich mesopores

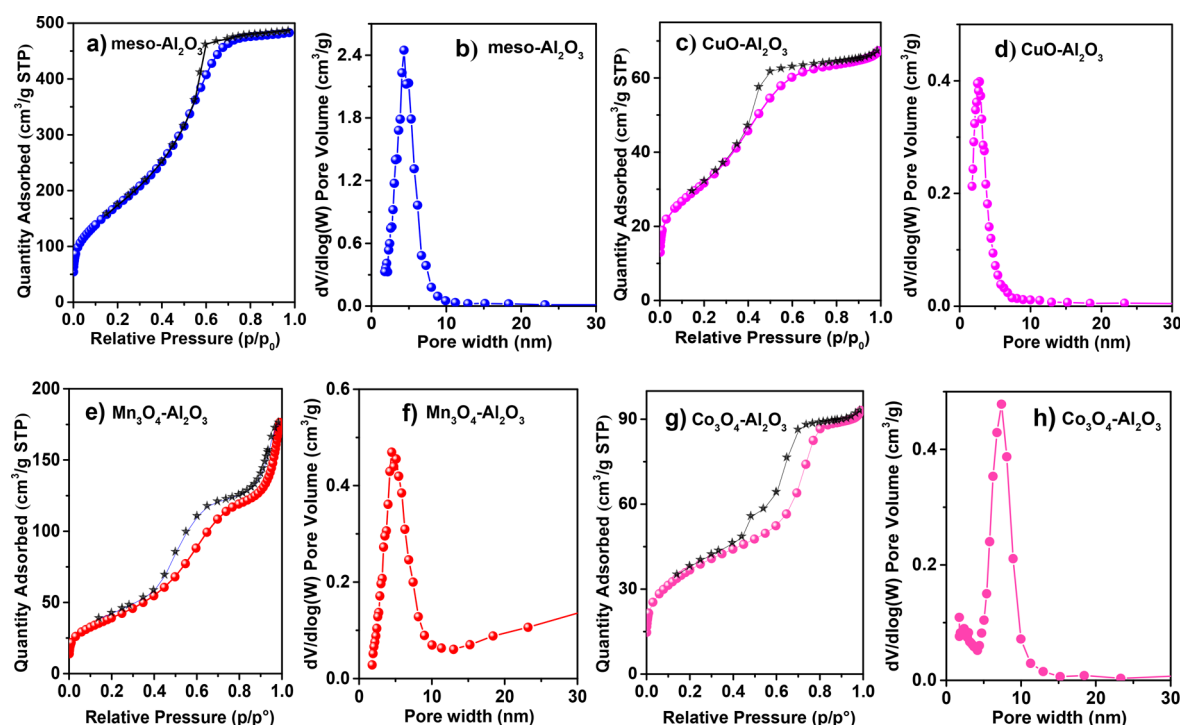


Figure 1. N_2 adsorption–desorption isotherms (a,c,e,g) and the corresponding pore size distribution (b,d,f,h) of meso- Al_2O_3 , $CuO-Al_2O_3$, $Mn_3O_4-Al_2O_3$, and $Co_3O_4-Al_2O_3$ synthesized using a mechanochemical NHSG method.

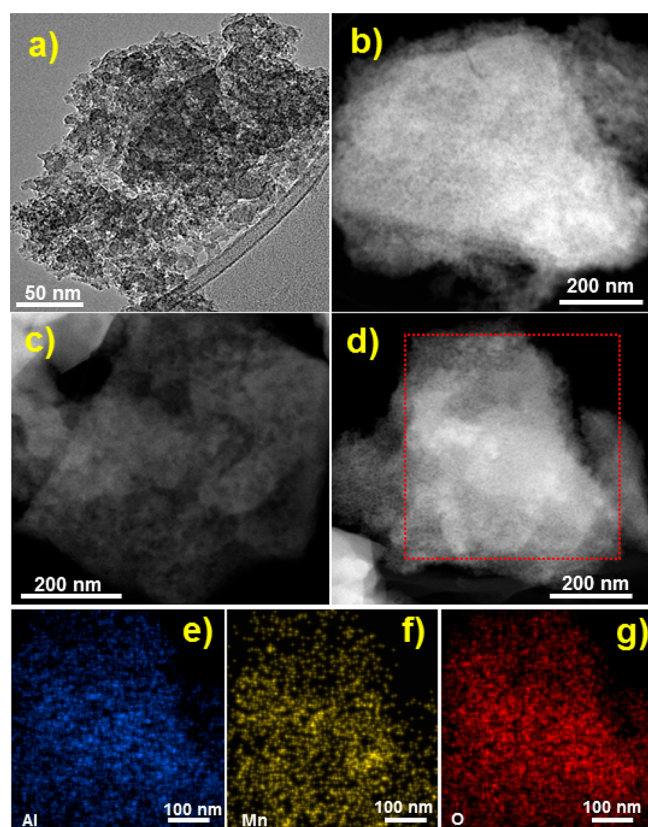


Figure 2. TEM a) and STEM-HAADF b) images of meso- Al_2O_3 ; STEM-HAADF images (c,d) of $Mn_3O_4-Al_2O_3$; and the corresponding element mapping signals e) Al, f) Mn and g) O.

Furthermore, energy-dispersive X-ray spectroscopy (EDS) mapping results of representative binary $Mn_3O_4-Al_2O_3$ sample (Figures 2e–g and Figure S9) show the well distribution of Mn, Al, and O, suggesting that the mesoporous structure is fabricated by the uniform mixture of Mn_3O_4 and Al_2O_3 . Additionally, TEM and EDS-mapping results for $CuO-Al_2O_3$ and $Co_3O_4-Al_2O_3$ samples in Figures S10 and S11 also display existence of mesopores and uniform mixture of CuO (Co_3O_4) and Al_2O_3 . Interestingly, the XRD result of $(CuNiFeCo)_x-Al_2O_3$ sample synthesized by the same method does not show characteristic diffraction peaks of CuO , Fe_2O_3 , or Co_3O_4 (Figure S12). The XRD pattern of $(CuNiFeCo)_x-Al_2O_3$ reveals its crystalline structure, which is close to 37.2 (111), 43.3 (012), 62.8 (110), and 75.4 (113) reflections of cubic NiO (JCPDS 65-2901). The above results suggest that Cu, Fe, and Co may be incorporated into the sublattice of NiO to form a solid solution. STEM-HAADF images of $(CuNiFeCo)_x-Al_2O_3$ sample reveal the aggregations of hybrid crystallites with a high degree of interstitial porosity (Figure 4a,b). EDS-mapping results show the highly uniform distribution of four metal species (Cu, Ni, Fe, and Co), further suggesting the existence of $(CuNiFeCo)_x-Al_2O_3$ solid solution (Figure S13). The pore nature of $(CuNiFeCo)_x-Al_2O_3$ sample was then evaluated via N_2 adsorption–desorption measurement at 77 K. A type-IV isotherm with an H_3 type hysteresis loop between the relative P/P_0 of 0.4 to 0.7 can be observed, leading to a surface area of $129\text{ m}^2\text{ g}^{-1}$ (Table 1) with narrow pore size distribution ($\sim 4\text{ nm}$) (Figure 5a,b). The molar ratio of Al:(CuNiFeCo) obtained from ICP-AES result is 1.9, similar to 1.5 determined from EDS result in Figure S13.

Crystalline high-entropy metal oxides (normally containing five or more metal species), as a new class of material, have attracted great interest because of their unique structure and potential applications.^{51,52} To date, the state-of-art high-entropy metal oxides only possess very limited surface area (e.g., $< 30\text{ m}^2\text{ g}^{-1}$),^{53,54} because their porous structures would collapse

side by side are observed throughout the material backbone in the range of several hundred nanometers (Figure 2c,d).

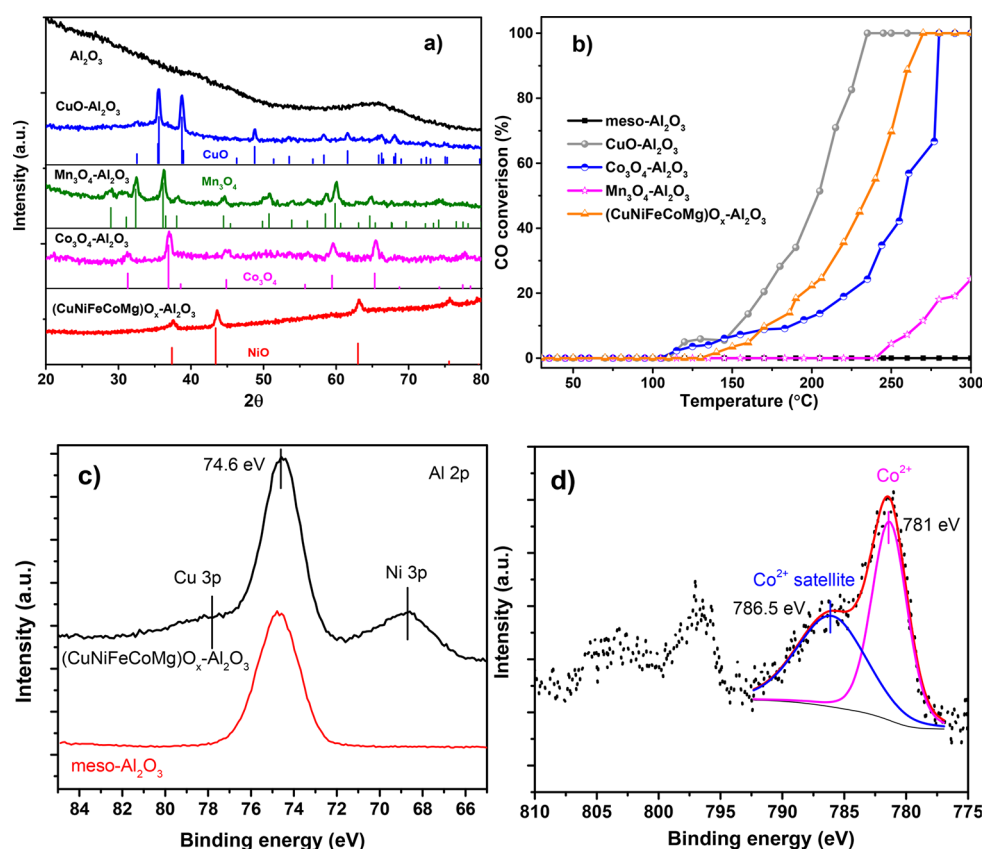


Figure 3. XRD results of a) meso-Al₂O₃, binary metal oxides and (CuNiFeCoMg)O_x-Al₂O₃; b) CO oxidation activity over different catalysts; normalized XPS spectra of c) Al 2p and d) Co 2p transitions for meso-Al₂O₃ (peak intensity for Al 2p is reduced) and (CuNiFeCoMg)O_x-Al₂O₃ samples.

during the removal of templates and the crystallization process, which limits their performance in catalysis. To synthesize mesoporous high entropy metal–Al mixed oxides, Mg was chosen as the fifth metal element for the preparation of (CuNiFeCoMg)O_x-Al₂O₃. The basis for the choice of these metals is their similar atomic radius. The corresponding XRD patterns are displayed (Figure 3a). It is reassuring that when Mg is used as the fifth metal, the diffraction peaks become similar to the shift of diffraction peaks of cubic crystalline NiO. The diffraction peaks belonging to CuO, Fe₂O₃, Co₃O₄, and Mn₃O₄ in binary metal–Al oxides disappeared accidentally, suggesting the formation of single high-entropy phase). The crystalline size of high-entropy phase is approximately 11.6 nm determined by Scherrer equation. Moreover, SEM-EDS results reveal that five elements (Cu, Ni, Fe, Co, and Mg) signal are simultaneous discovered in selected regions (Figure 4e–l). The obtained results suggested that the high-entropy metal oxides (CuNiFeCoMg)O_x should be formed and mixed uniformly with Al₂O₃ to form a mesoporous complex. We deduced that the mechanochemical process together with calcination step in are the major reason for the formation of solid solution.

The metal valence is illustrated by X-ray photoelectron spectroscopy (XPS) analysis, and the results are shown in Figures 3c,d and S14. The Al 2p XPS spectra in Figure 3c show the primary peak located at 74.6 eV for both meso-Al₂O₃ and (CuNiFeCoMg)O_x-Al₂O₃ samples, suggesting the existence of Al–O bonds in Al³⁺ (often known as Al₂O₃ species). The Cu 2p spectrum in Figure S14a can be fitted into four peaks, corresponding to Cu 2p_{3/2} (933.0 eV) and Cu 2p_{1/2} (953.0 eV) and their satellite peaks. These characteristics match well

with standard peaks of CuO. The similar results in Ni 2p spectrum can identify the possible existence of NiO (Figure S14b).⁵⁵ The main peak from Mg 2p spectrum in Figure S14c is located around 50 eV, but the valence state of Mg is difficult to determine due to the close peak position of different state (Mg⁰ and Mg²⁺). However, Mg should exist in MgO form considering that the synthesis process is difficult to generate Mg⁰. For Co 2p spectrum in Figure 3d, the peaks at 781 eV together with strong satellite at 786.5 eV indicate the existence of Co²⁺, since the signature of Co³⁺ have no satellite peak.⁵⁶ The Fe 2p spectrum in Figure S14d suggests the coexistence of Fe²⁺ and Fe³⁺.

STEM-HAADF images of (CuNiFeCoMg)O_x-Al₂O₃ sample suggest the existence of rich mesoporosity marked with small yellow circles (Figure 4c,d), which is also revealed by SEM images in Figure S15. The pore nature of (CuNiFeCoMg)O_x-Al₂O₃ sample was then evaluated by N₂ adsorption–desorption measurement at 77 K. A type-IV isotherm with steep adsorption peaks between the relative pressure (*P*/*P*₀) of 0.2–0.8, revealing the abundant mesoporous structure (Figure 5c,d). It offered a surface area up to 198 m² g^{−1}, and the relatively lower surface area compared to meso-Al₂O₃ should be attributed to the blockage and collapse of pore channels during the crystallization process of high-entropy metal oxides. The molar ratio of Al: (CuNiFeCoMg) obtained from the ICP-AES result is 1.3.

Subsequently, the catalytic activities of different mesoporous metal oxides are evaluated for CO oxidation (Figure 3b). The *T*₁₀₀ values (temperature for 100% CO conversion) are 220 °C (CuO–Al₂O₃), 270 °C (Co₃O₄–Al₂O₃), and 260 °C ((CuNiFeCoMg)O_x-Al₂O₃), respectively. Among these catalysts, Mn₃O₄-Al₂O₃ exhibits much worse activity for the

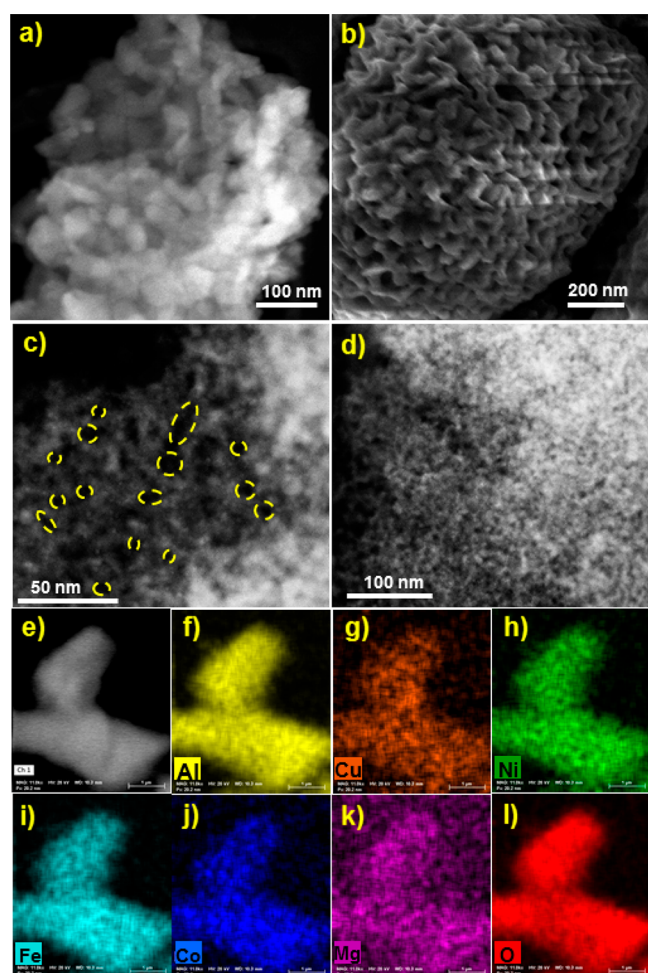


Figure 4. STEM (a) and SEM (b) images of $(\text{CuNiFeCo})\text{O}_x\text{-Al}_2\text{O}_3$; STEM (c,d) images of $(\text{CuNiFeCoMg})\text{O}_x\text{-Al}_2\text{O}_3$; SEM-EDS image (e) of $(\text{CuNiFeCoMg})\text{O}_x\text{-Al}_2\text{O}_3$; and the corresponding element mapping signals f) Al, g) Cu, h) Ni, (i) Fe, j) Co, k) Mg, and l) O.

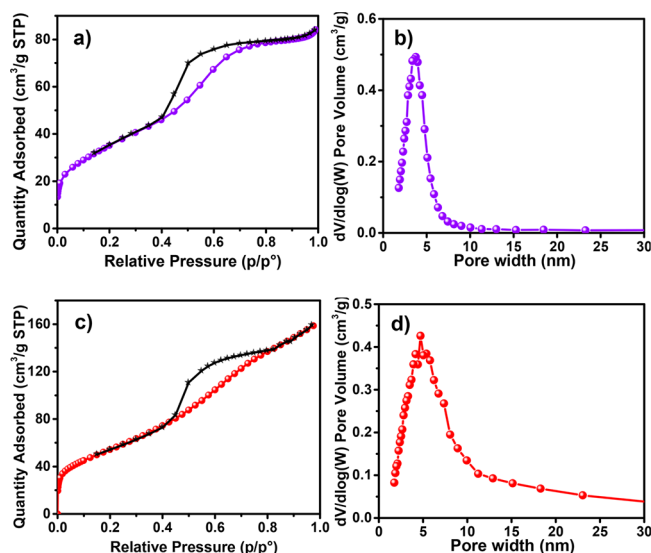


Figure 5. N_2 adsorption–desorption isotherms and the corresponding pore size distribution of $(\text{CuNiFeCo})\text{O}_x\text{-Al}_2\text{O}_3$ (a,b) and $(\text{CuNiFeCoMg})\text{O}_x\text{-Al}_2\text{O}_3$ (c,d) synthesized using mechanochemical NHSG method.

oxidation of CO, which may be attributed to the relatively inactive of Mn_3O_4 phase. And the mesoporous $\text{CuO-Al}_2\text{O}_3$ catalyst becomes active at 120°C , and CO was completely converted at 220°C , which are superior to most catalysts especially for Cu–Al mixed oxides prepared using traditional impregnation and precipitation method as shown in Table S2.^{57–59} Moreover, both $\text{CuO-Al}_2\text{O}_3$ and $(\text{CuNiFeCoMg})\text{O}_x\text{-Al}_2\text{O}_3$ catalysts exhibited remarkable stability even after being used for 48 h (Figure S16a,b). Considering the facile, fast, and solvent-free preparation method, these mesoporous mixed oxides may have great potential applications in industrial catalysis. As shown in Figure S17, XRD results of fresh and spent $\text{CuO-Al}_2\text{O}_3$ and $(\text{CuNiFeCoMg})\text{O}_x\text{-Al}_2\text{O}_3$ catalysts display similar diffraction peaks and crystallite size. Additionally, temperature-programmed desorption (TPD) of CO in Figure S18 reveals similar CO desorption peak area, indicating the dispersion of metal active species does not decrease remarkably after use.

The sulfur tolerance is of great importance in real-world catalysis. However, CO oxidation activity of most catalysts went down remarkably in the presence of SO_2 , greatly affecting the process of industrialization. Therefore, we compared the performance of SO_2 resistance over $\text{CuO-Al}_2\text{O}_3$ and $(\text{CuNiFeCoMg})\text{O}_x\text{-Al}_2\text{O}_3$ catalysts. Before the catalytic activity test, each catalyst was treated in 1000 ppm of SO_2 at 280°C for 5 h. The catalytic activity of $\text{CuO-Al}_2\text{O}_3$ decreased remarkably after SO_2 treatment (Figure S19a). It is noteworthy that there is a negligible effect of SO_2 on $(\text{CuNiFeCoMg})\text{O}_x\text{-Al}_2\text{O}_3$ catalyst (Figure S19b). The decreased activity of $\text{CuO-Al}_2\text{O}_3$ is attributed to the formation of sulfites, evidenced by the feature in FTIR spectra around 1100 cm^{-1} (Figure S10).^{60,61} However, it is clear that the formation of crystalline high-entropy metal oxides protect itself from SO_2 poisoning, contributing to the superior SO_2 tolerance compared with $\text{CuO-Al}_2\text{O}_3$. This result indicates that the $(\text{CuNiFeCoMg})\text{O}_x\text{-Al}_2\text{O}_3$ catalyst may be stabilized by the relative higher configurational entropy, preventing itself from being sulfurated.⁵¹

4. CONCLUSIONS

In summary, an efficient, fast, and facile route for the synthesis of mesoporous Al_2O_3 and its metal oxide hybrids has been developed based on a mechanochemical nonhydrolytic method. The obtained meso- Al_2O_3 shows a record high surface area ($\sim 644\text{ m}^2\text{ g}^{-1}$) and uniform pore size ($\sim 5\text{ nm}$). Moreover, mesoporous metal–aluminum oxide and four or five metal solid solution–aluminum oxide hybrids with rich mesoporous structure and single cubic crystalline phase was discovered. More importantly, this mechanochemical method does not need solvents and can shorten the pore filling process from 1–3 days to 60 min. The obtained mesoporous mixed metal oxide (such as $(\text{CuNiFeCoMg})\text{O}_x\text{-Al}_2\text{O}_3$) exhibit not only an exceptional catalytic performance in CO oxidation, but also the superior SO_2 tolerance. This mechanochemical NHSG method may open the door to the scalable preparation of a new class of metal oxide catalysts.

■ ASSOCIATED CONTENT

Supporting Information

The Supporting Information is available free of charge on the ACS Publications website at DOI: 10.1021/acs.chemmater.9b01244.

Additional results of characterization results (XRD, SEM, TEM, N₂ adsorption–desorption, CO-TPD, ICP-AES, and FTIR) (PDF)

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Notes

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