



# Exploring Extreme Environments with Nanometer Resolution

K. Hattar

Sandia National Laboratories

March 23<sup>rd</sup>, 2018



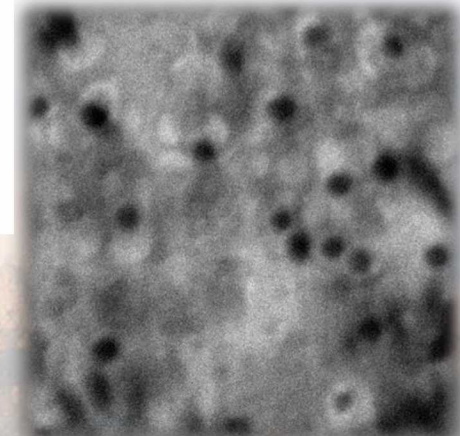
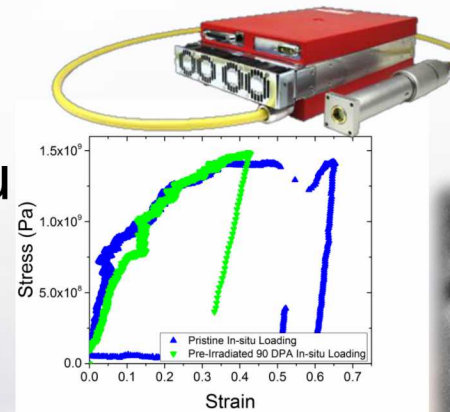
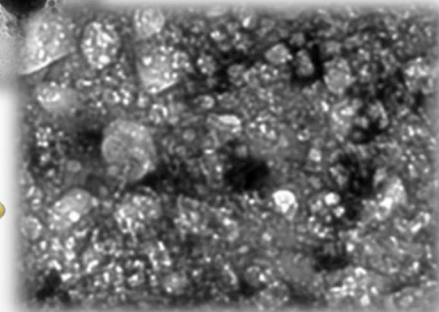
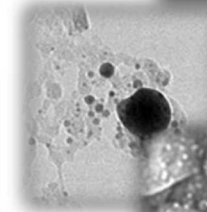
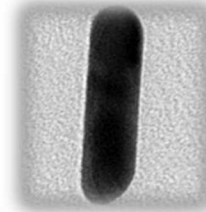
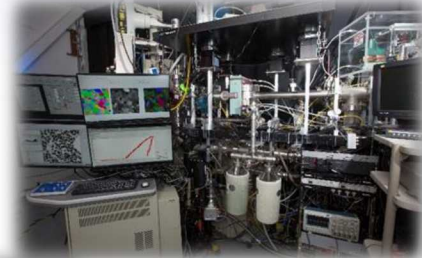
## Collaborators:

- D.L. Buller, D.C. Bufford, S.H. Pratt, T.J. Boyle, B.A. Hernandez-Sanchez, S.J. Blair, B. Muntifering, C. Chisholm, P. Hosemann, A. Minor, J. A. Hinks, F. Hibberd, A. Ilinov, D. C. Bufford, F. Djurabekova, G. Greaves, A. Kuronen, S. E. Donnelly, K. Nordlund, F. Abdeljawad, S.M. Foiles, J. Qu, C. Taylor, J. Sugar, P. Price, C.M. Barr, D. Adams, M. Abere, L. Treadwell, A. Cook, A. Monterrosa, IDES Inc, J. Sharon, B. L. Boyce, C. Chisholm, H. Bei, E.P. George, W. Mook, Hysitron Inc., G.S. Jawaharram, S. Dillon, R.S. Averback, N. Heckman, J. Carroll, S. Briggs, E. Carnes, J. Brinker, D. Sasaki, T. Nenoff, B.G. Clark, P.J. Cappillino, B.W. Jacobs, M.A. Hekmaty, D.B. Robinson, L.R. Parent, I. Arslan, & Protochips, Inc.

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# Outline

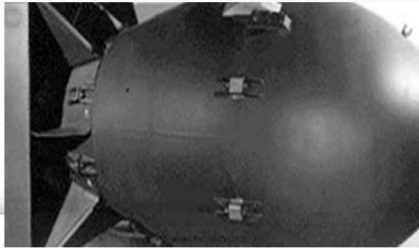
1. Introduction to Sandia National Laboratories and its *In situ* Ion Radiation Transmission Electron Microscope (I<sup>3</sup>TEM)
2. Understanding radiation stability in nanoparticles
3. Effect of radiation environments on nanocrystalline metals
4. Lasers in a TEM (why not?)
5. State-of-art in quantitative in situ mechanical testing
6. Other environments (in situ SEM, liquid, and gas)





# Sandia National Laboratories

*"Exceptional service in the national interest"*



May 13, 1949

Dear Mr. Wilson:

I am informed that the Atomic Energy Commission intends to ask that the Bell Telephone Laboratories accept under contract the direction of the Sandia laboratory at Albuquerque, New Mexico.

This operation, which is a vital segment of the atomic weapons program, is of extreme importance and urgency in the national defense, and should have the best possible technical direction.

I hope that after you have heard more in detail from the Atomic Energy Commission, your organization will find it possible to undertake this task. In my opinion you have here an opportunity to render an exceptional service in the national interest.

I am writing a similar note direct to Dr. C. E. Buckley.

Very sincerely yours,

Mr. Leroy A. Wilson,  
President,  
American Telephone and Telegraph Company,  
195 Broadway,  
New York 7, N. Y.



Livermore, CA



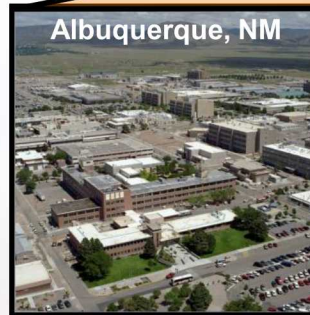
Nevada Test Site



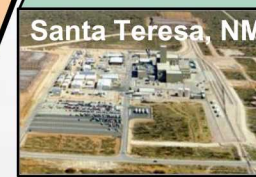
Yucca Mountain



Kauai



Albuquerque, NM



Santa Teresa, NM



Pantex, TX



SANDIA NATIONAL LABORATORIES  
President Harry S. Truman Fellowship in  
National Security Science and Engineering

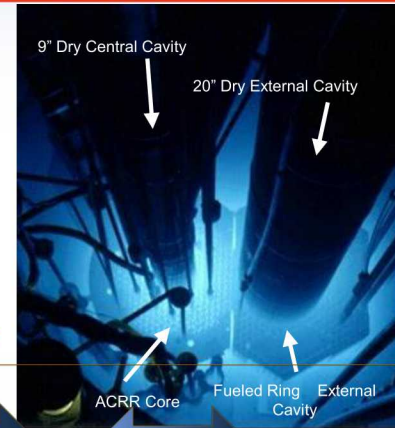
**"Sandia develops advanced technologies to ensure global peace."  
– S. Younger**



Sandia National Laboratories

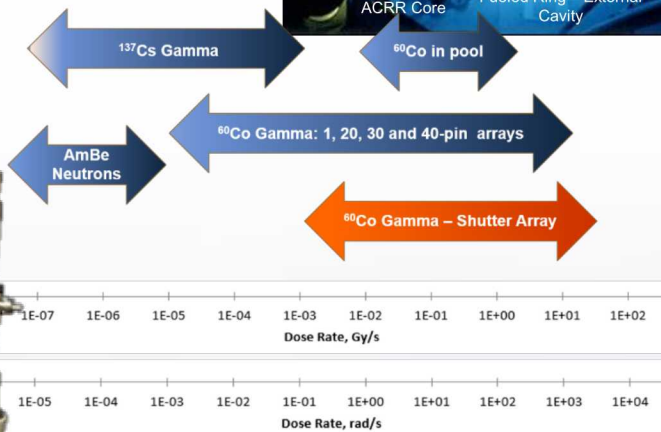
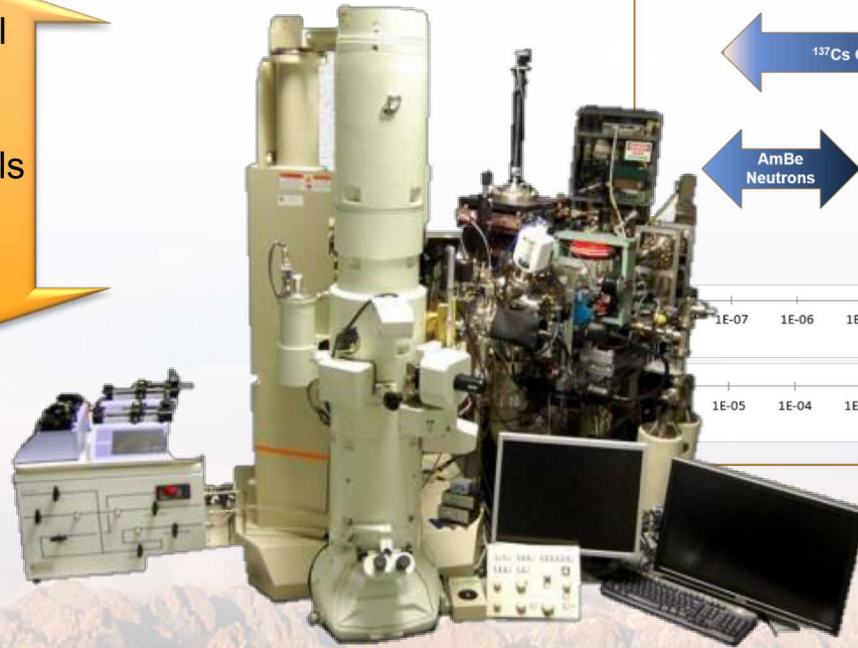
# Sandia's USER Capabilities

## Core Facility - SNL

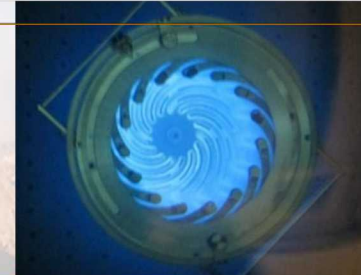


D. Hanson, W. Martin, M. Wasiolek

- Nanophotonics & Optical Nanomaterials
- Soft- Biological & Composite Nanomaterials
- Quantum Materials
- In-situ Characterization and Nanomechanics



## Gateway Facility - LANL

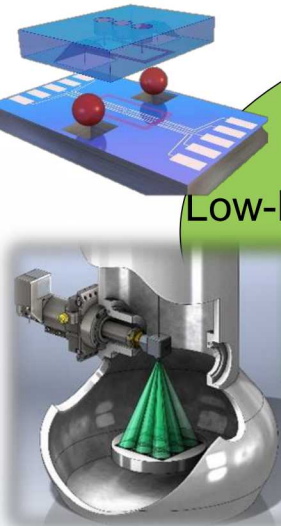


Sandia National Laboratories is a multi-mission laboratory managed and operated by National Technology and Engineering Solutions of Sandia, LLC., a wholly owned subsidiary of Honeywell International, Inc., for the U.S. Department of Energy's National Nuclear Security Administration under contract DE-NA0003525.



# *In situ* Characterization and Nanomechanics

Developing and implementing world-leading capabilities to study the dynamic response of materials and nanosystems to mechanical, electrical, or other stimuli.



## In-Situ Characterization

Optical Microscopy

Low-Energy Electron Microscopy

Scanning Electron Microscopy

Transmission Electron Microscopy

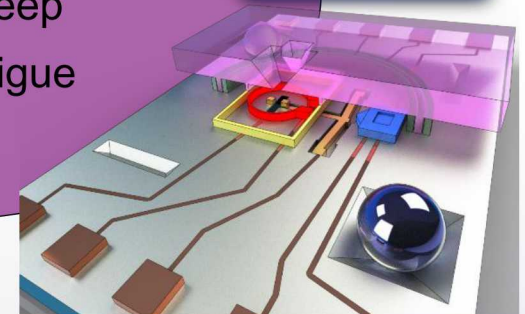
## In-Situ Nanomechanics

Indentation

Monotonic Loading

Creep

Fatigue



## Scientific challenges

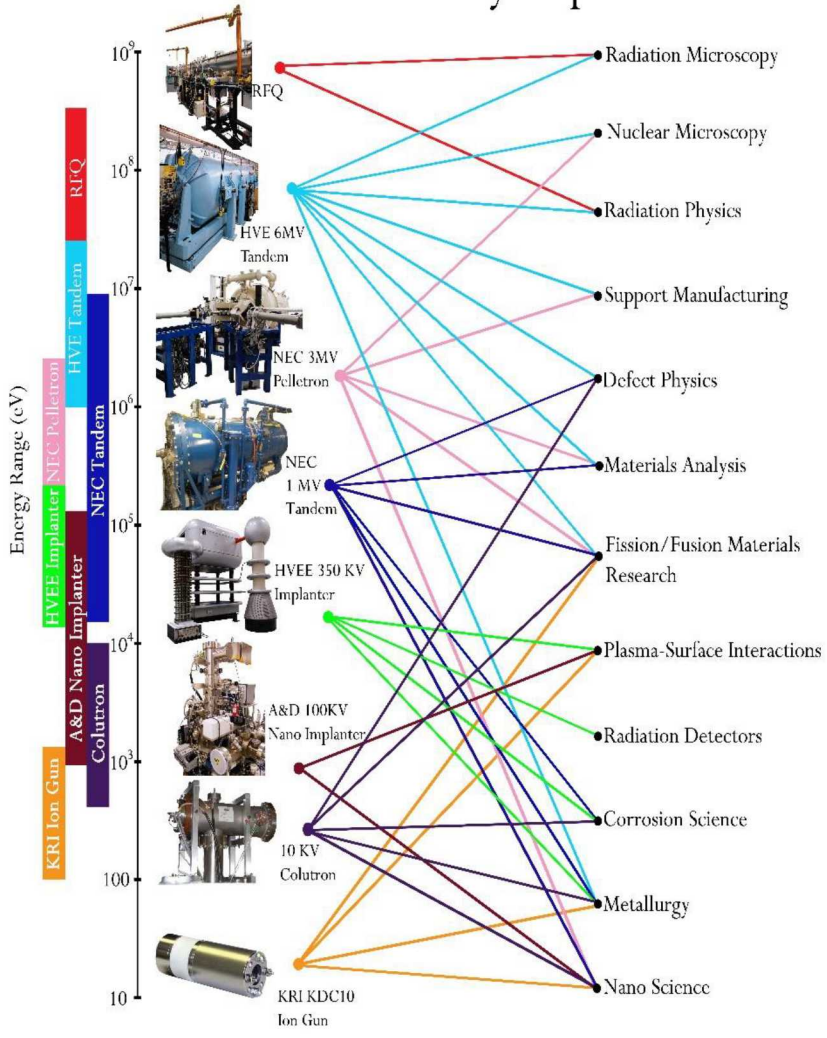
- How do defects and crystal distortions alter the mechanical and other extrinsic properties in nanostructured materials?
- How can we understand and control energy transfer across interfaces and over multiple length and time scales?
- How does the environment change the mechanical response and surface structure of nanoscale materials?



# Sandia's Ion Beam Laboratory



## Ion Beam Laboratory Capabilities

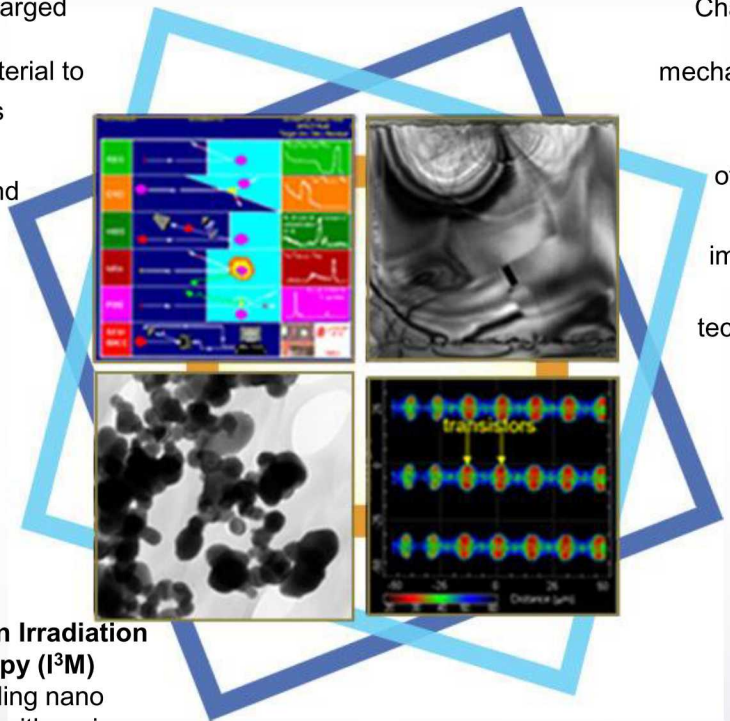


### Ion Beam Analysis (IBA)

Shooting a charged particle at an unknown material to determine its identity, local chemistry, and structure.

### Ion Beam Modification (IBM)

Changing the optical, mechanical, and chemical properties of materials via ion implantation to meet technological needs



### In Situ Ion Irradiation Microscopy (I<sup>3</sup>M)

Bombarding nano samples with various particles and observing the changes in real time to understand how materials will behave in extreme environments.

### Radiation Effects Microscopy (REM)

Using ion emissions to determine the Radiation hardness of microelectronics, identifying potential weaknesses.

**The IBL has a unique and comprehensive ion beam capability set including and *In situ* Ion Irradiation Transmission Electron Microscopy.**

# Benefits & Limitations of *in situ* TEM

## Benefits

1. Real-time nanoscale resolution observations of microstructural dynamics

## Limitations

1. Predominantly limited to microstructural characterization
  - Some work in thermal, optical, and mechanical properties
2. Limited to electron transparent films
  - Can often prefer surface mechanisms to bulk mechanisms
  - Local stresses state in the sample is difficult to predict
3. Electron beam effects
  - Radiolysis and Knock-on Damage
4. Vacuum conditions
  - $10^{-7}$  Torr limits gas and liquid experiments feasibility
5. Local probing
  - Portions of the world study is small

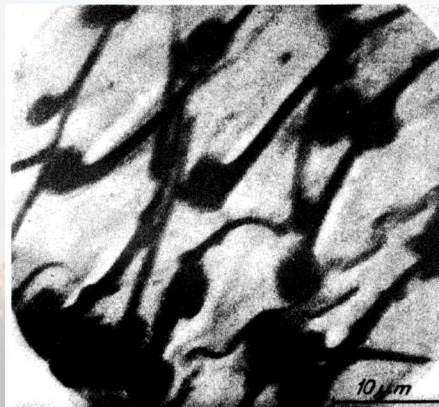


Fig. 6: Wing surface of the house fly.  
 (First internal photography,  $U = 60$  kV,  $M_x = 2200$ )  
 (Driest, E., and Müller, H.O. Z. Wiss. Mikroskope 52, 53-57 (1955))

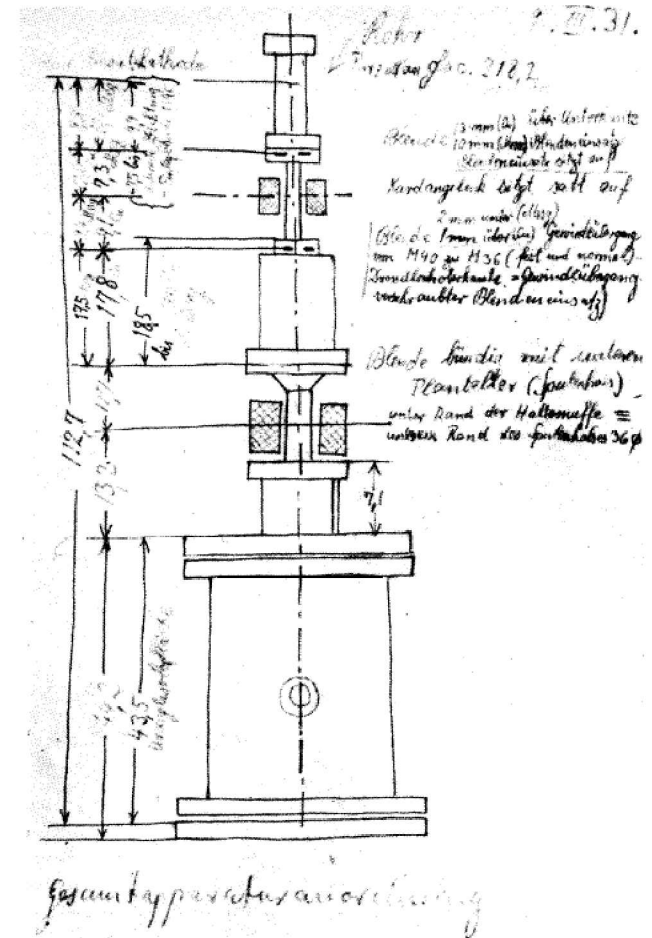


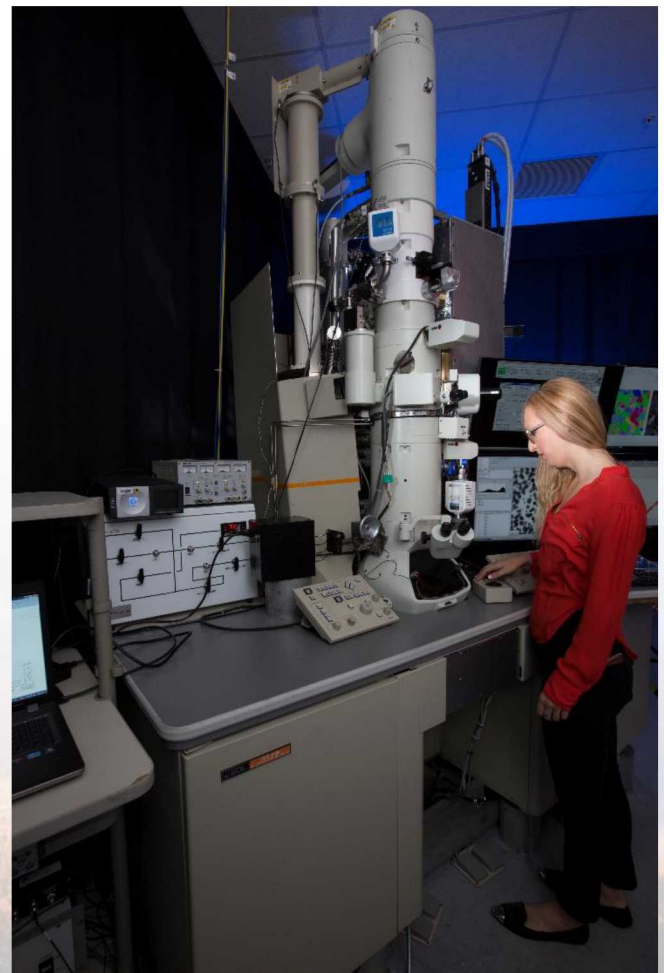
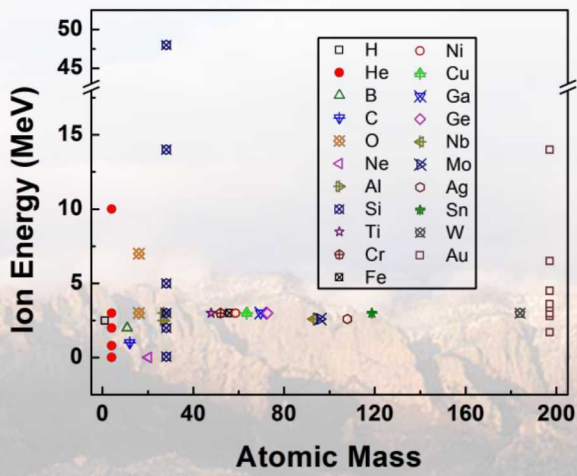
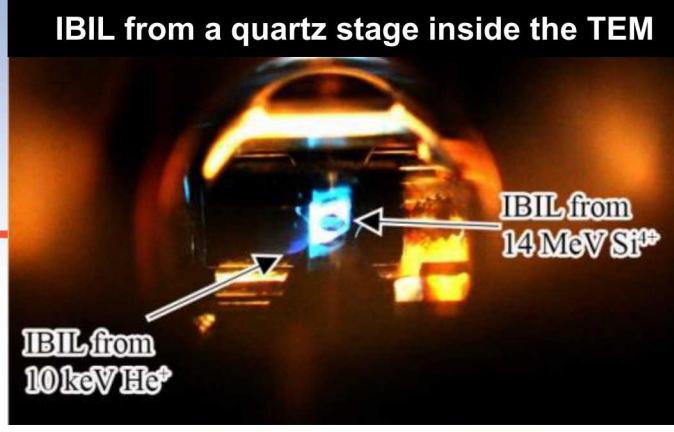
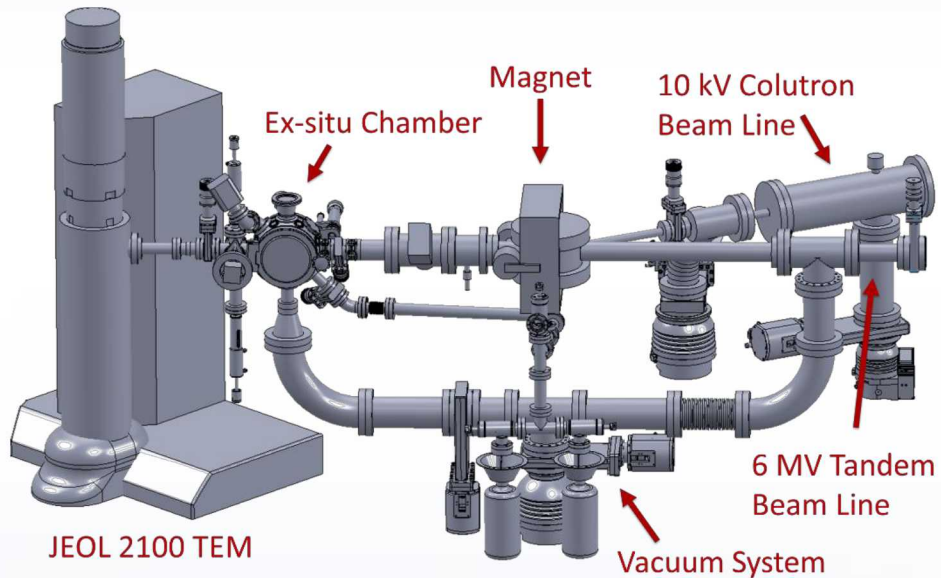
Fig. 2: Sketch by the author (9 March 1931) of the cathode ray tube for testing one-stage and two-stage electron-optical imaging by means of two magnetic electron lenses (electron microscope) [8].



# Sandia's Concurrent *In situ* Ion Irradiation TEM Facility

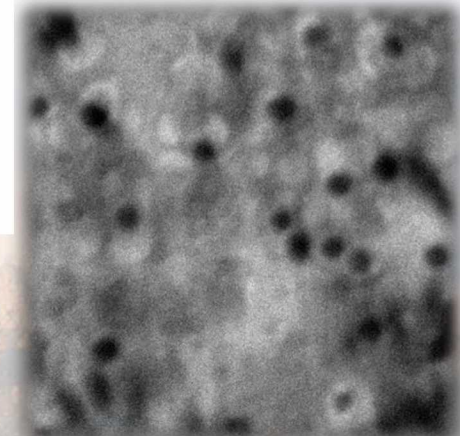
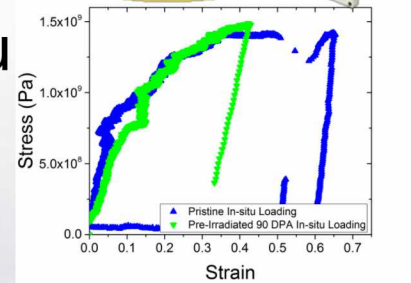
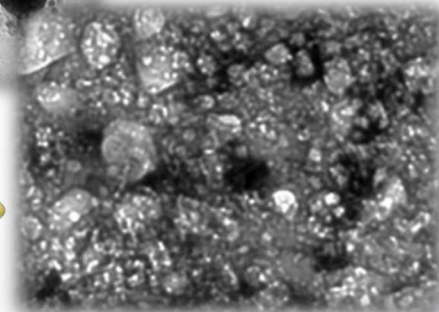
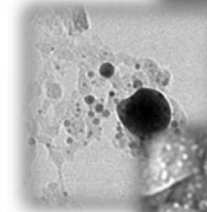
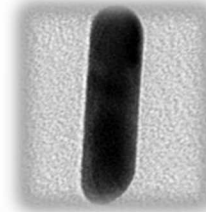
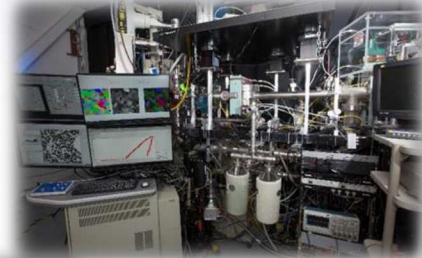
Collaborator: D.L. Buller

10 kV Colutron - 200 kV TEM - 6 MV Tandem



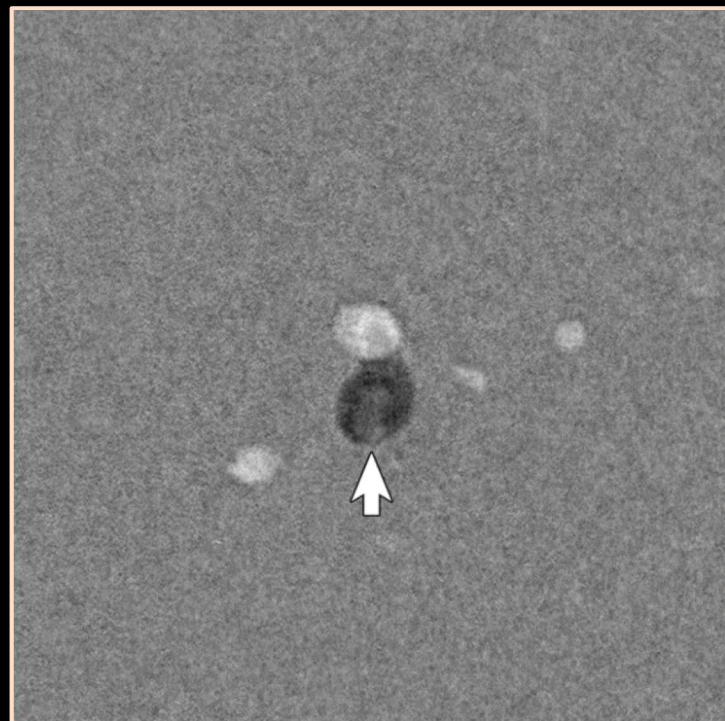
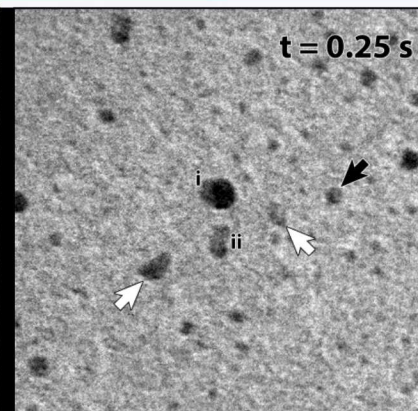
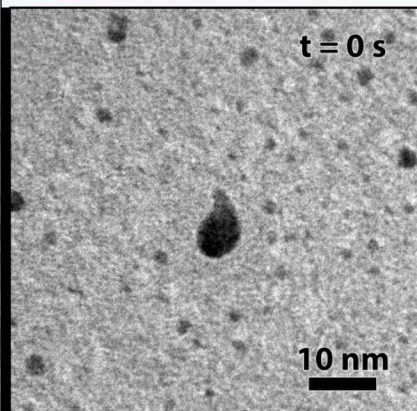
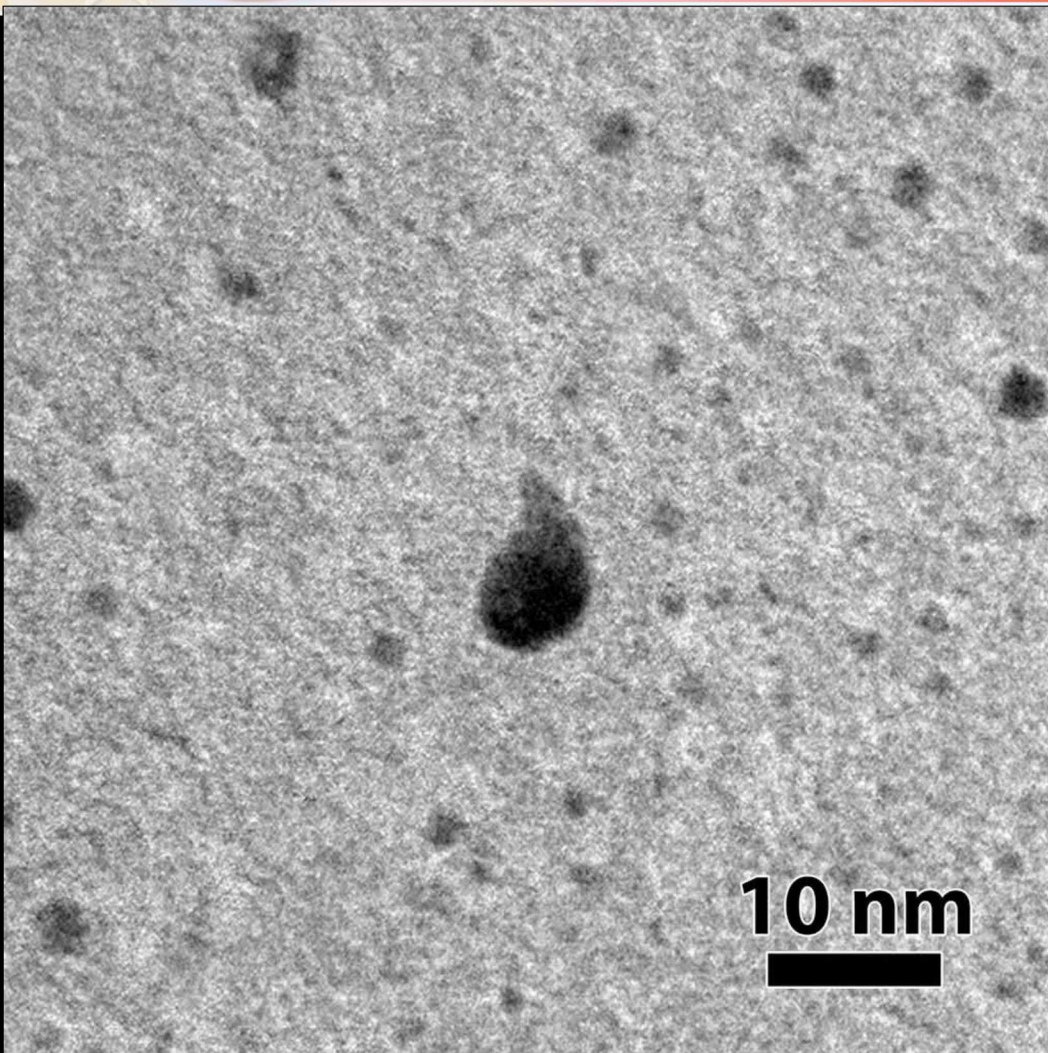
# Outline

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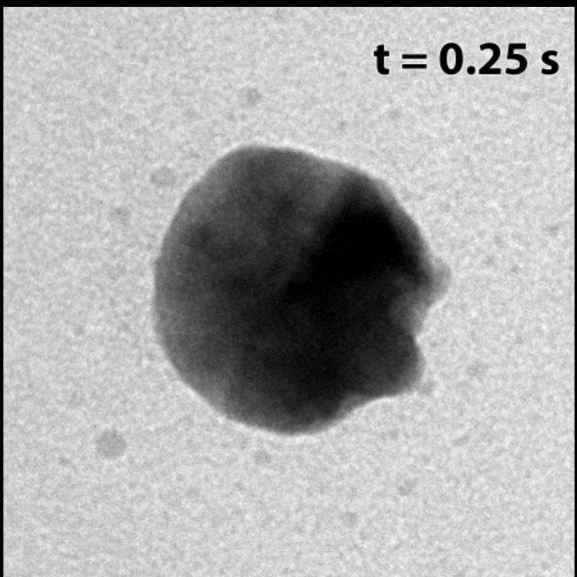
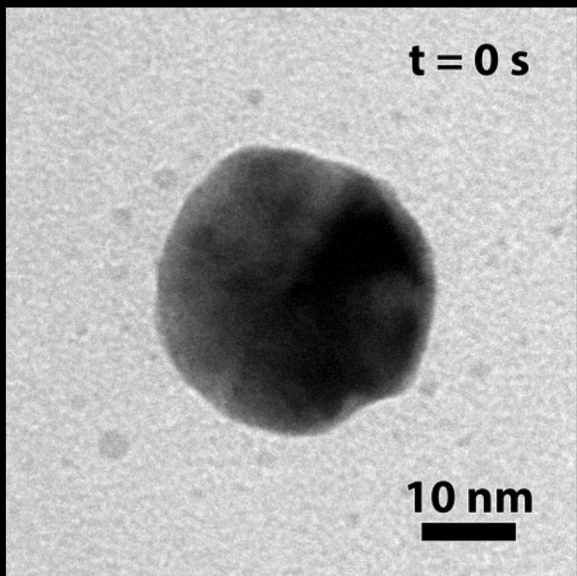
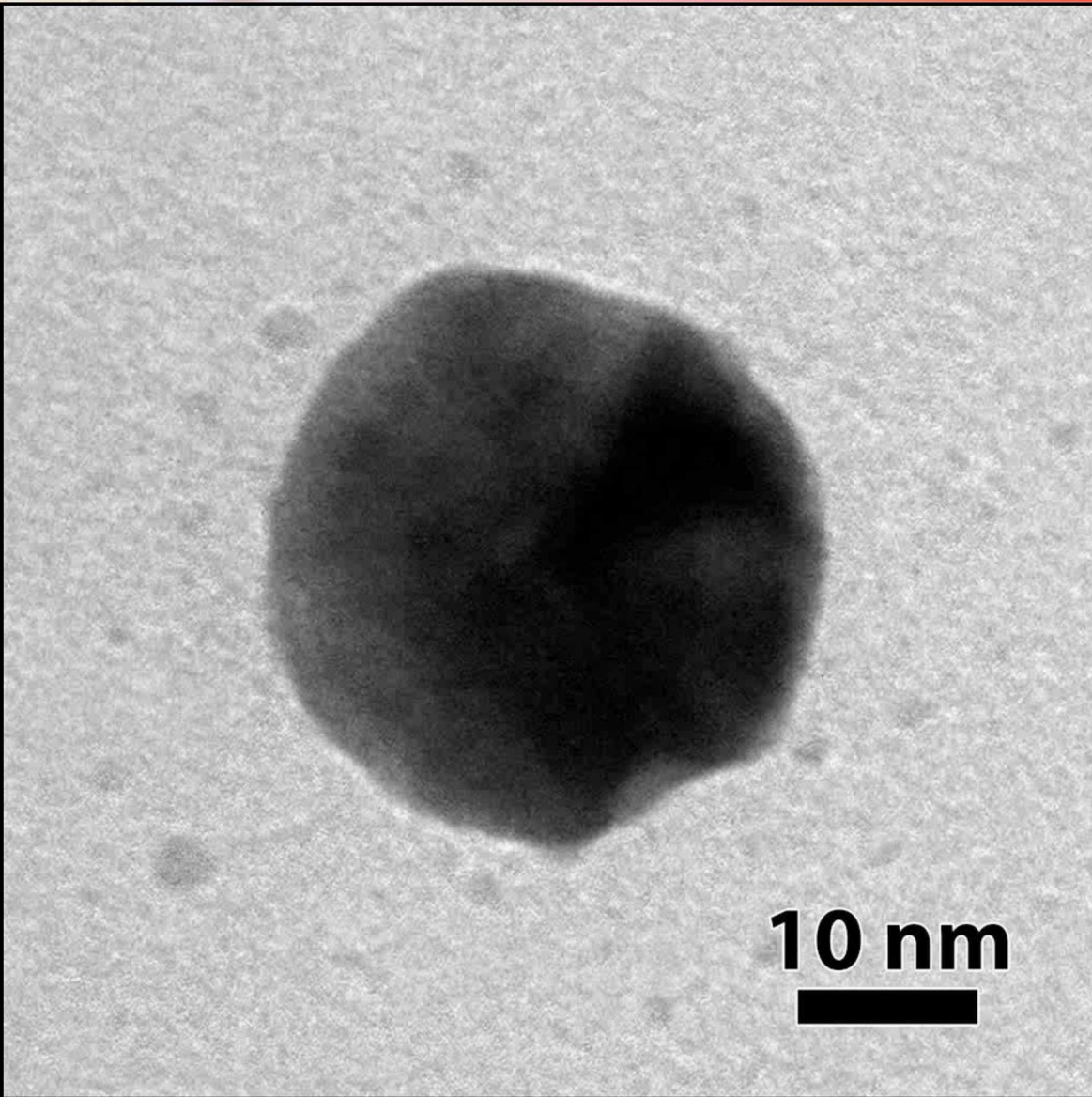
# Single Ion Strikes: 46 keV Au<sup>1+</sup> ions into 5 nm Au nanoparticles

Collaborator: D.C. Bufford



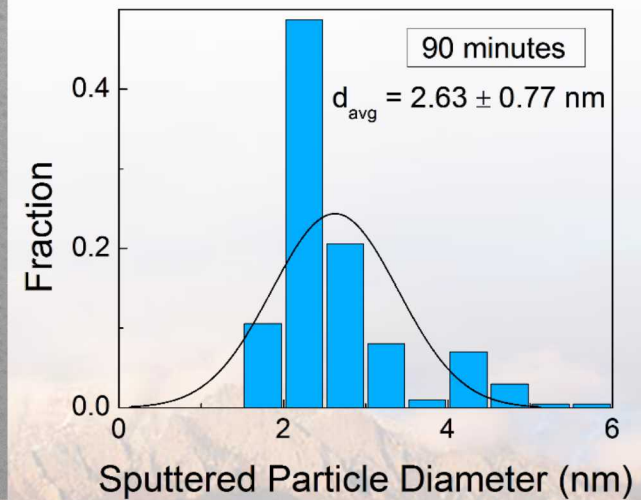
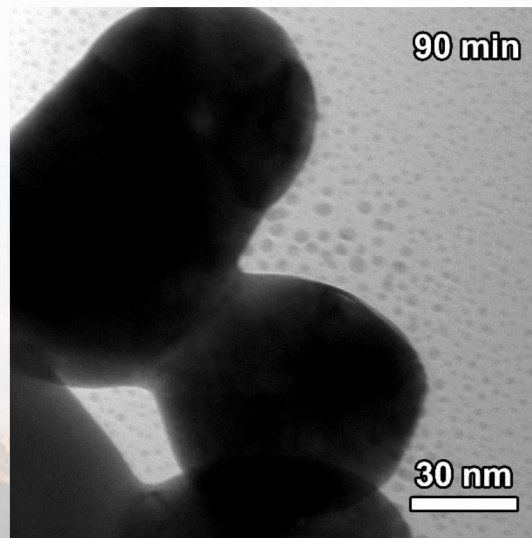
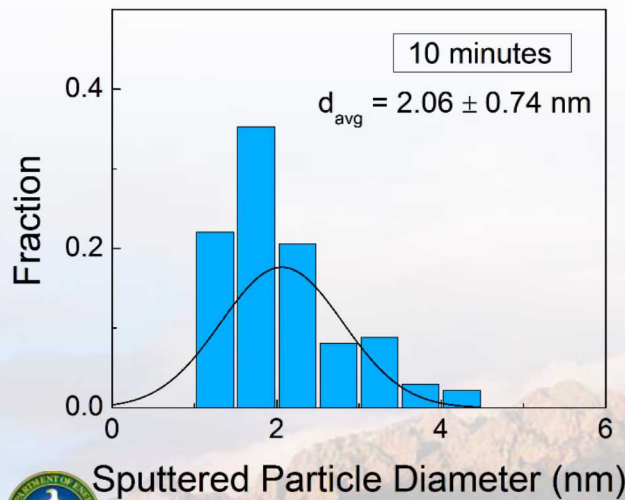
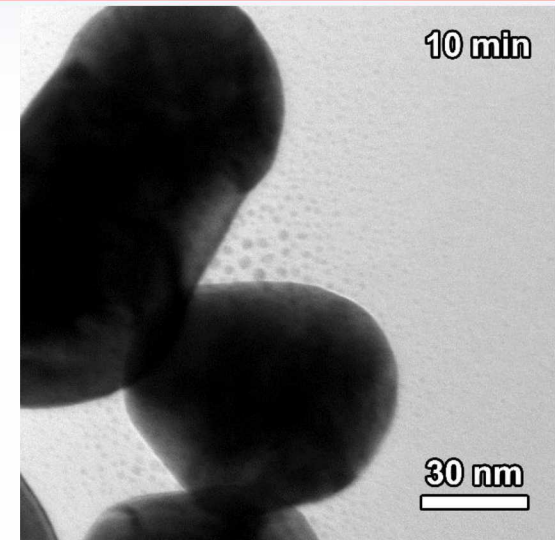
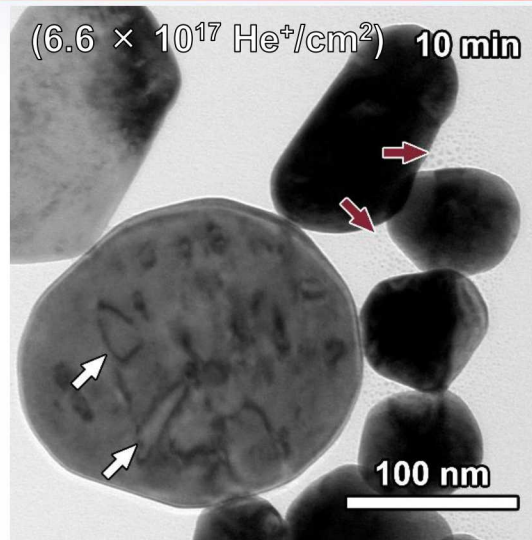
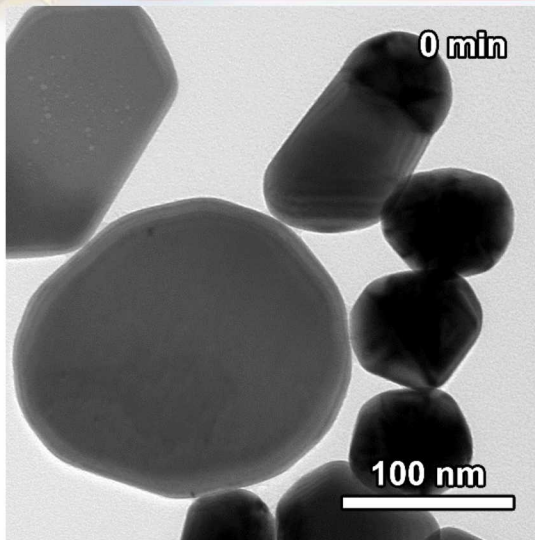
# Single Ion Strikes: 46 keV Au<sup>1+</sup> ions into 20 nm NPs

Collaborator: D.C. Bufford



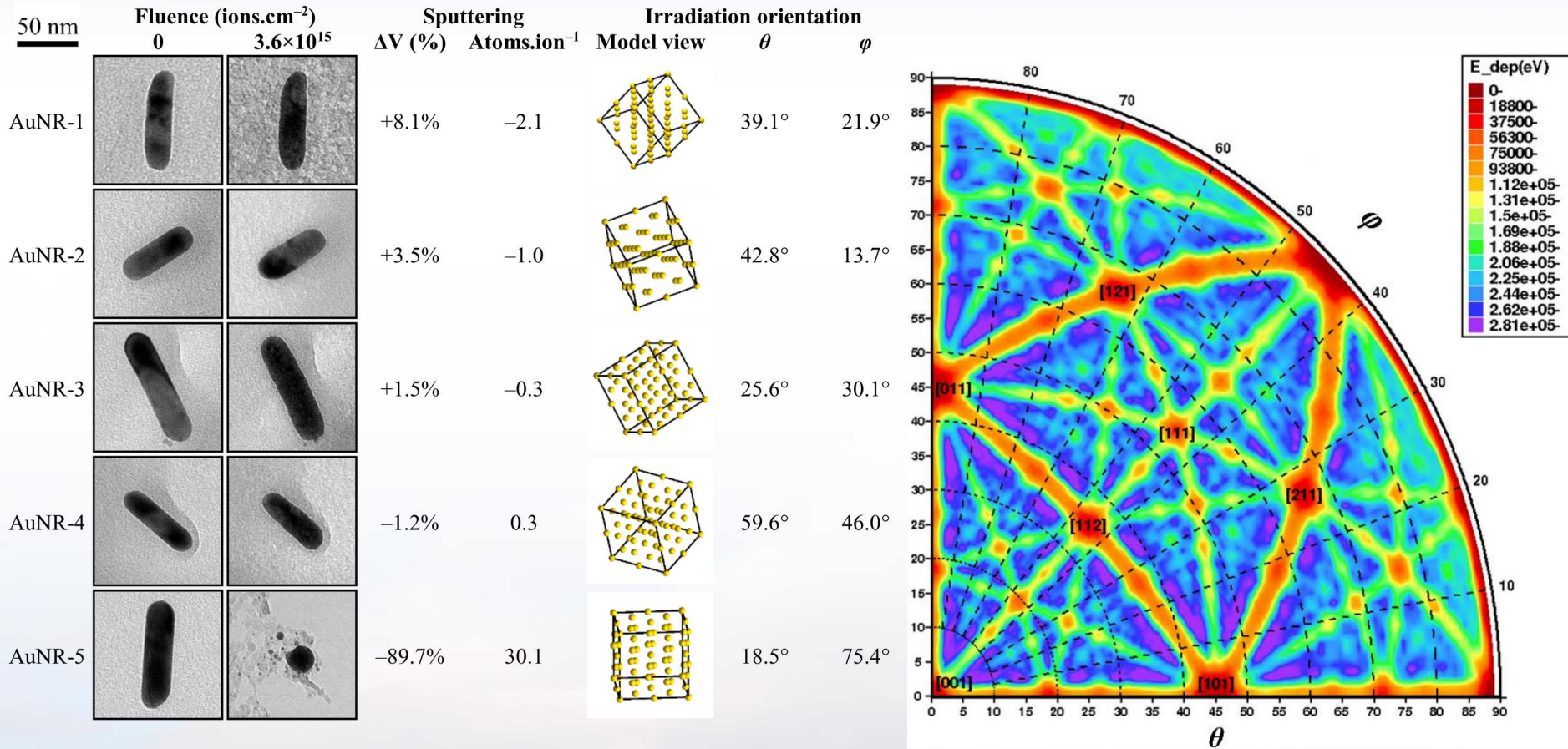
# Formation of Dislocation Loops & Sputtered Particles due to He implantation

Collaborators: D.C. Bufford, S.H. Pratt & T.J. Boyle



# Exploring Radiation effects in Au Nanorods

Collaborators: J. A. Hinks, F. Hibberd, A. Ilinov, D. C. Bufford, F. Djurabekova, G. Greaves, A. Kuronen, S. E. Donnelly & K. Nordlund



**Crystal Orientation Matters!**



# Radiation Tolerance is Needed in Advanced Scintillators for Non-proliferation Applications

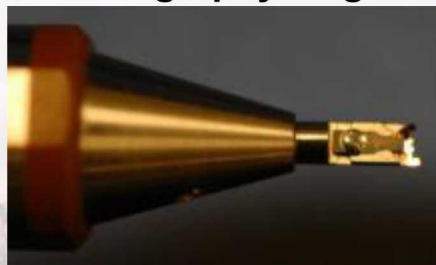
Contributors: S.M. Hoppe, B.A. Hernandez-Sanchez, T. Boyle



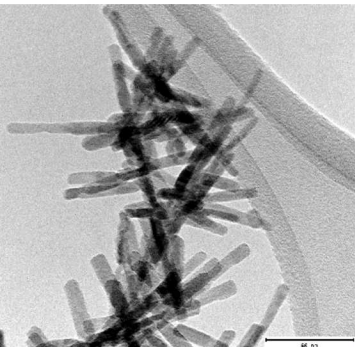
## *In situ* Ion Irradiation TEM (I<sup>3</sup>TEM)



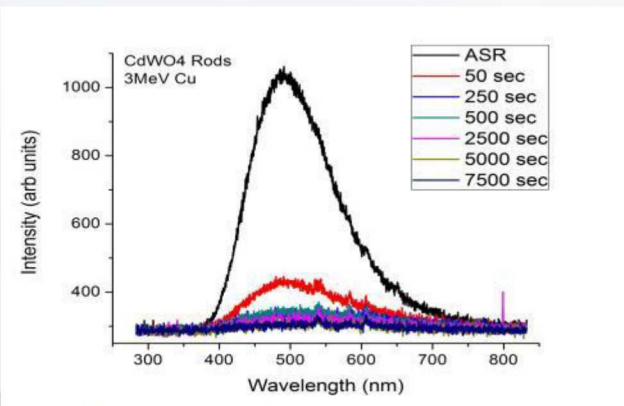
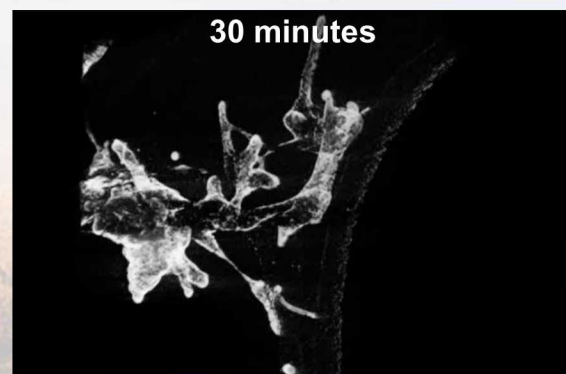
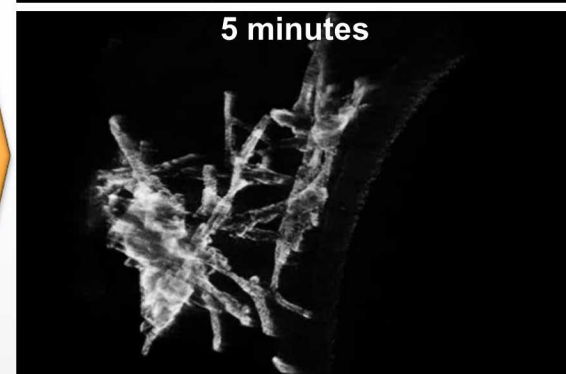
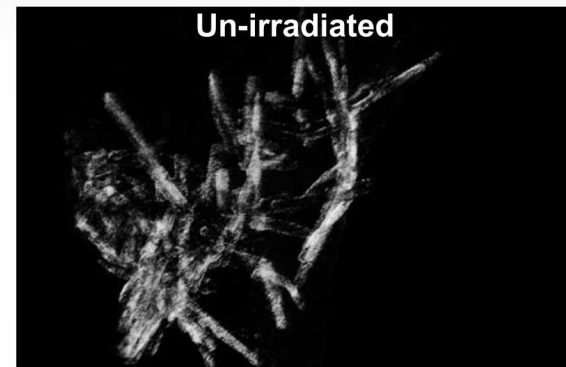
Hummingbird tomography stage



Tomography of Irradiated CdWO<sub>4</sub>:  
3 MeV Cu<sup>3+</sup> at ~30 nA

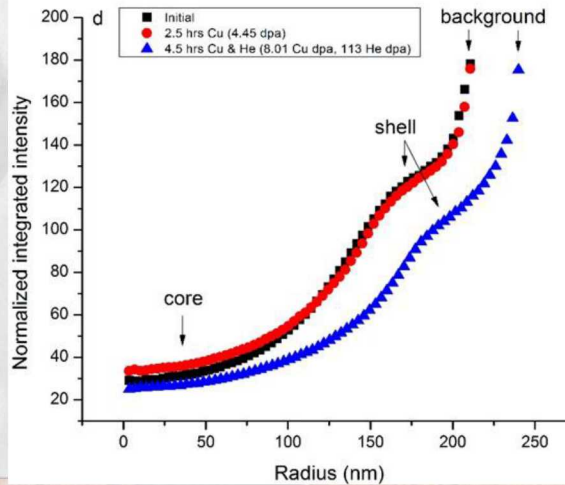
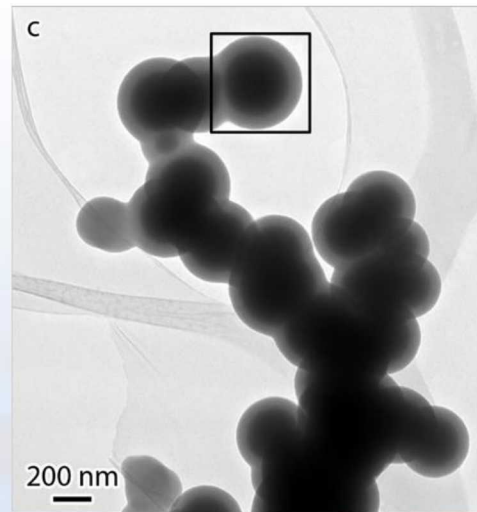
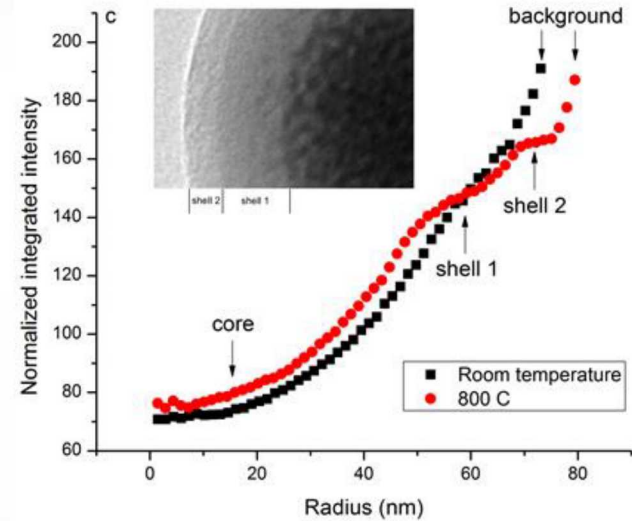
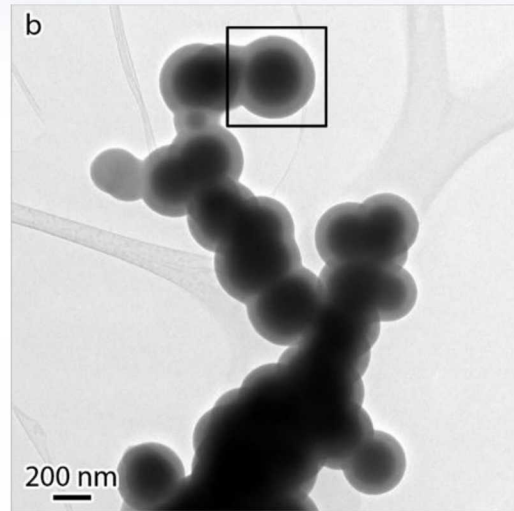
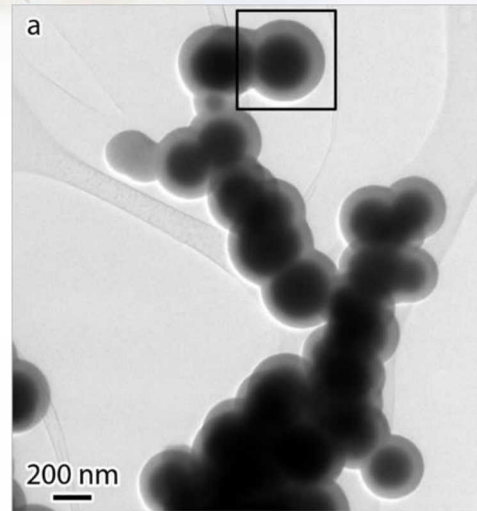


High-Z nanoparticles (CdWO<sub>4</sub>) are promising, but are radiation sensitive



# Radiation Stable Nanoparticles!

Collaborators: T.J. Boyle, S.J. Blair, B. Muntiferung



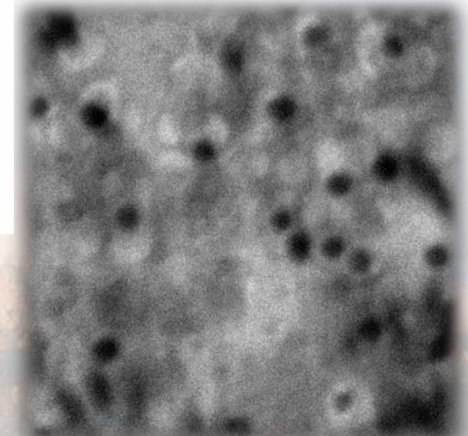
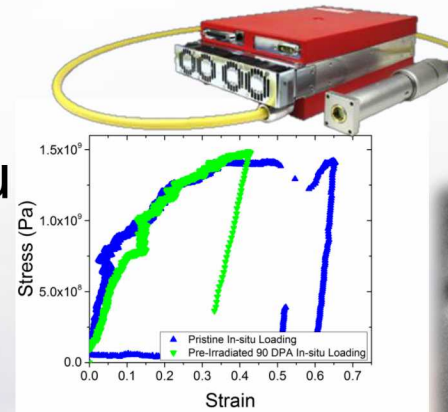
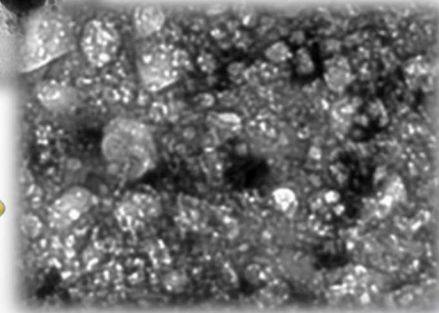
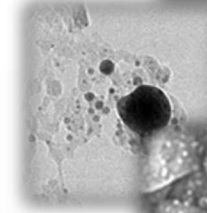
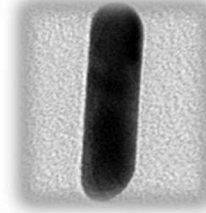
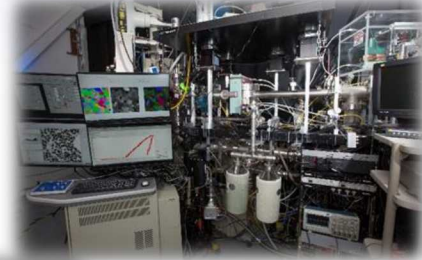
Ion	Energy (keV)	Dose (ions/cm <sup>2</sup> ) Dose rate (ions/cm <sup>2</sup> /s)	Damage (dpa) Damage rate (dpa/s)
Cu	3000	1.2 x 10 <sup>14</sup>	4.45
		1.3 x 10 <sup>10</sup>	4.9 x 10 <sup>-4</sup>
He	10	2.3 x 10 <sup>18</sup>	113
		1.4 x 10 <sup>14</sup>	7.0 x 10 <sup>-3</sup>
& Cu	3000	2.2 x 10 <sup>14</sup>	8.01
		1.3 x 10 <sup>10</sup>	4.9 x 10 <sup>-4</sup>



In contrast some NPs appear to be very radiation stable

# Outline

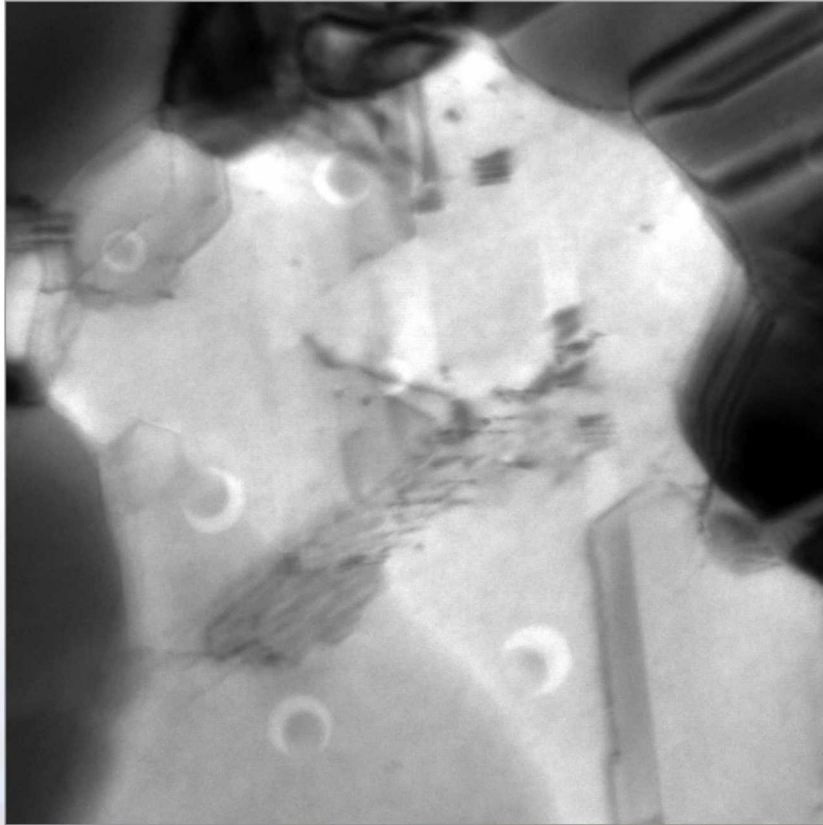
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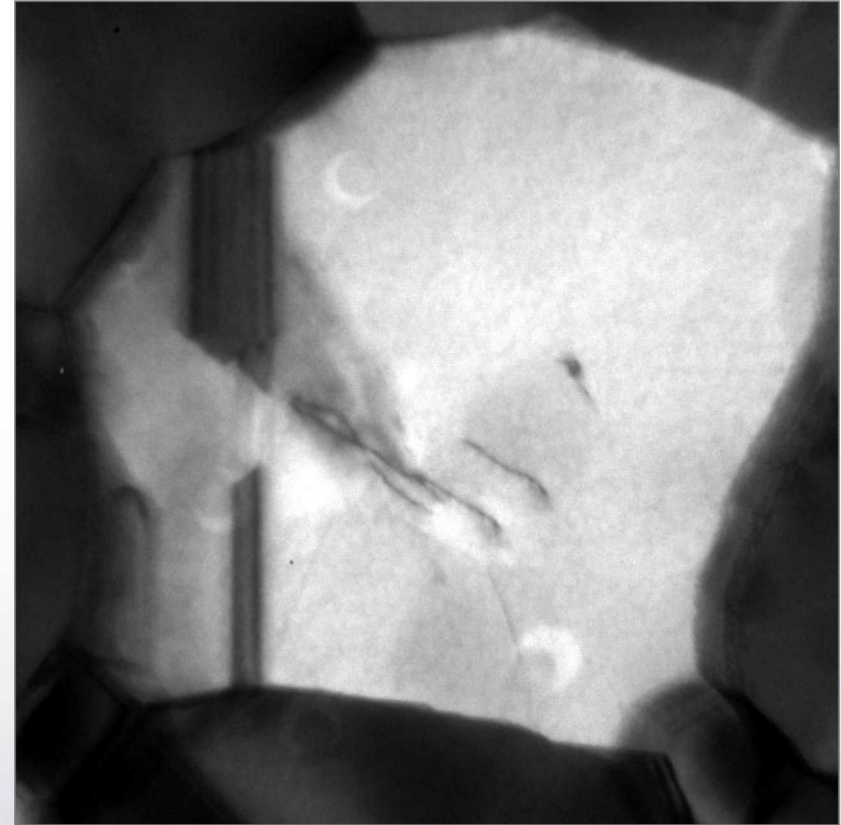
# Dose Rate Effects in Nanocrystalline Metals

Collaborators: C. Chisholm , P. Hosemann, & A. Minor

$7.9 \times 10^9$  ions/cm<sup>2</sup>/s



$6.7 \times 10^7$  ions/cm<sup>2</sup>/s



**VS**

Improved vibrational and ion beam stability permits us to work at 120kx or higher permitting imaging of single cascade events

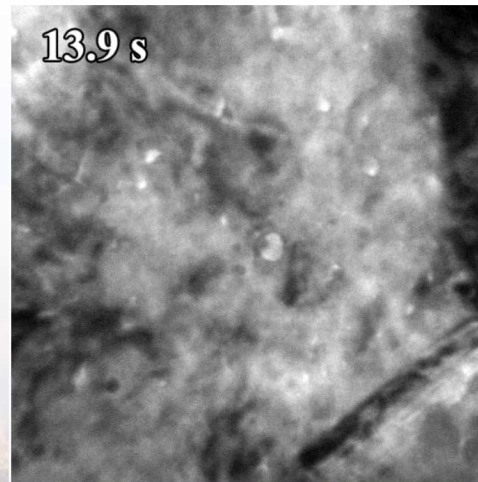
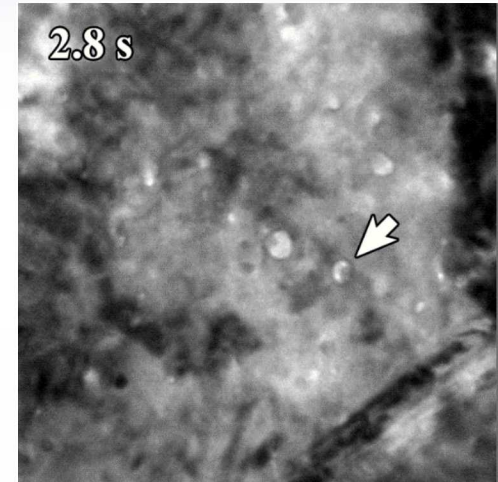
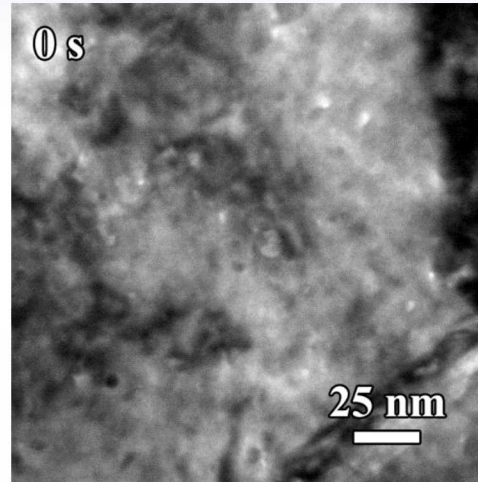
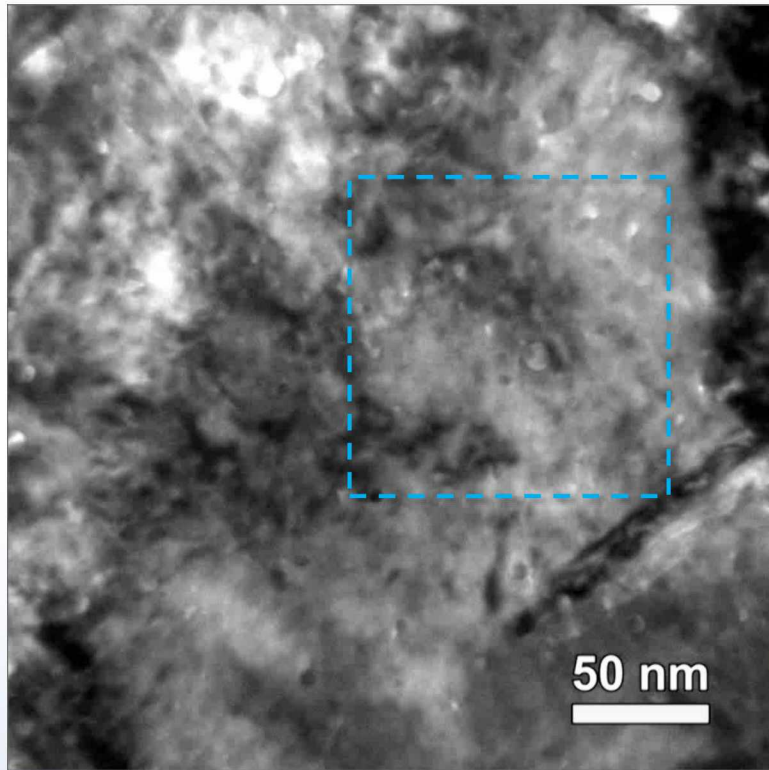


# Simultaneous *In situ* TEM Triple Beam:

## 2.8 MeV Au<sup>4+</sup> + 10 keV He<sup>+</sup> / D<sub>2</sub><sup>+</sup>

Collaborator: D.C. Bufford

Video playback speed x1.5.



In-situ triple beam He, D<sub>2</sub>, and Au beam irradiation has been demonstrated on Sandia's I<sup>3</sup>TEM!

Intensive work is still needed to understand the defect structure evolution that has been observed.

### ■ Approximate fluence:

- Au  $1.2 \times 10^{13}$  ions/cm<sup>2</sup>
- He  $1.3 \times 10^{15}$  ions/cm<sup>2</sup>
- D  $2.2 \times 10^{15}$  ions/cm<sup>2</sup>

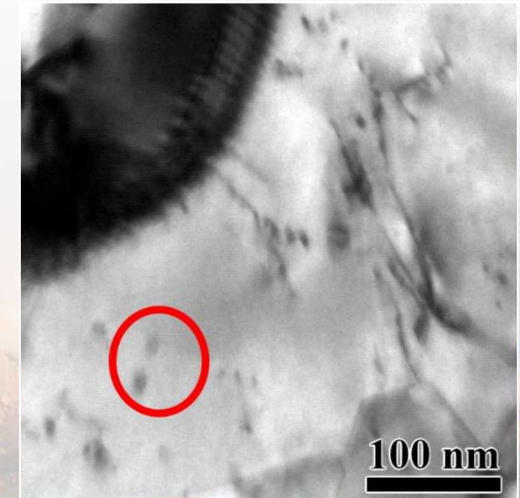
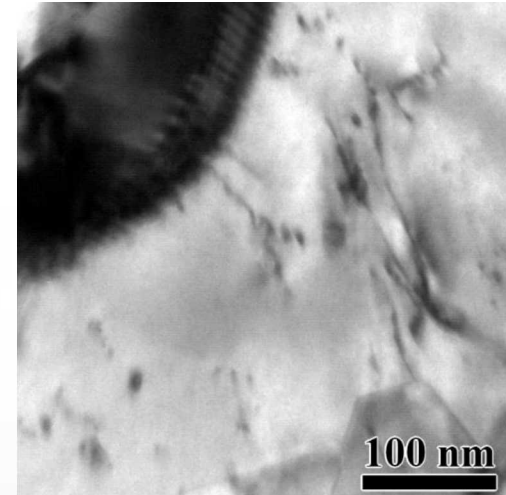
### ■ Cavity nucleation and disappearance



# 1D Brownian Motion in Real Time

Collaborator: D.C. Bufford

Triple beam condition:  
2.8 MeV Au<sup>4+</sup> + 10 keV He<sup>+</sup>/D<sub>2</sub><sup>+</sup>



100 nm

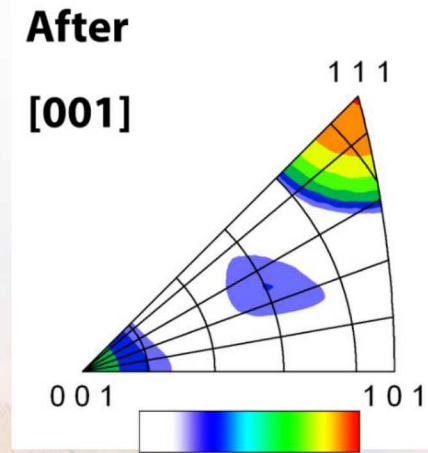
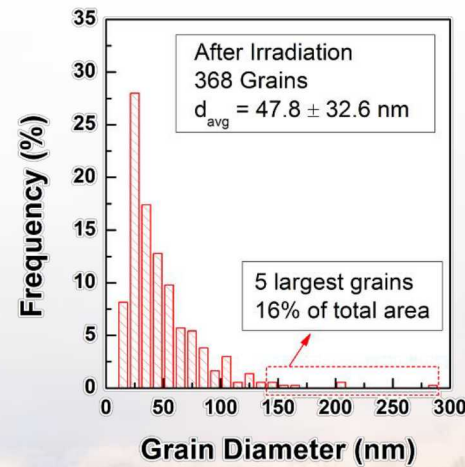
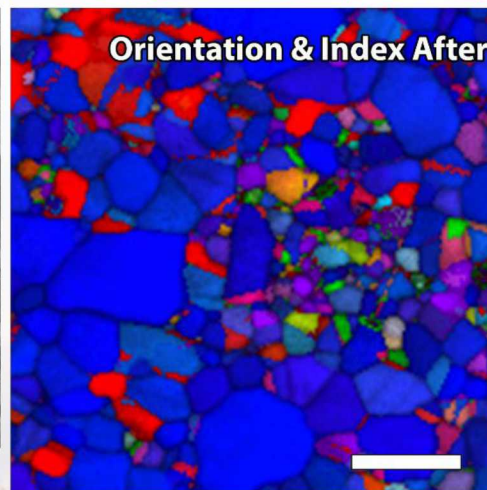
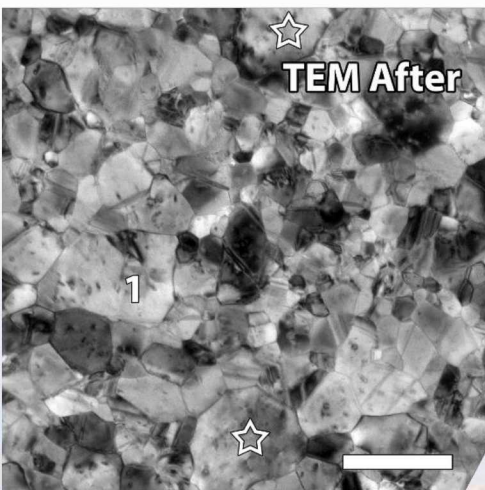
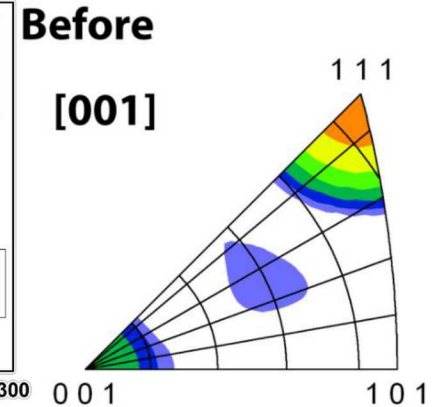
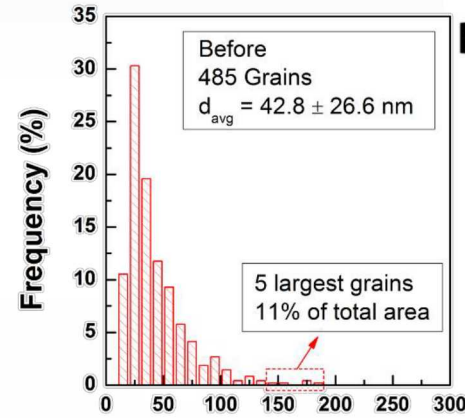
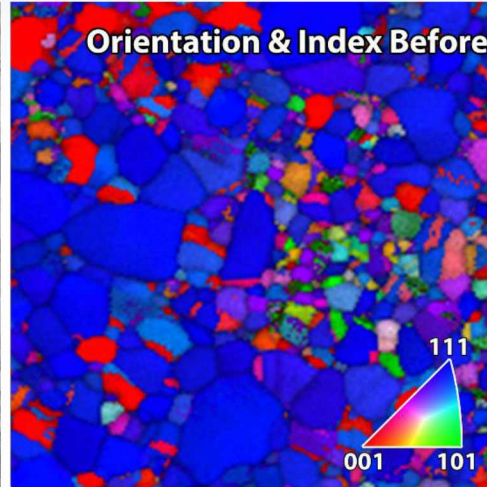
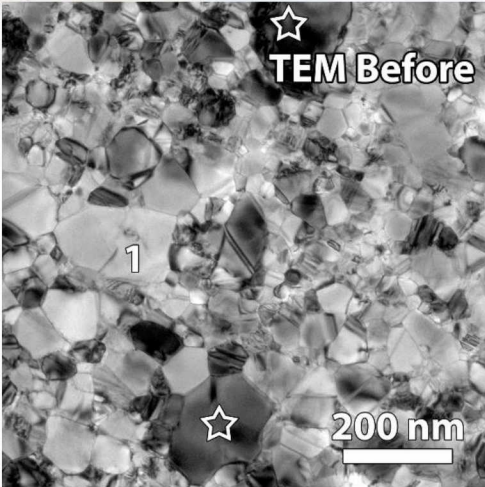
100 nm

- Dislocation loop moves between two pinning sites
  - ~30 nm apart



# Quantifying Grain Boundary Radiation Stability of Nanocrystalline Au

Collaborators: D.C. Bufford, F. Abdeljawad, & S.M. Foiles



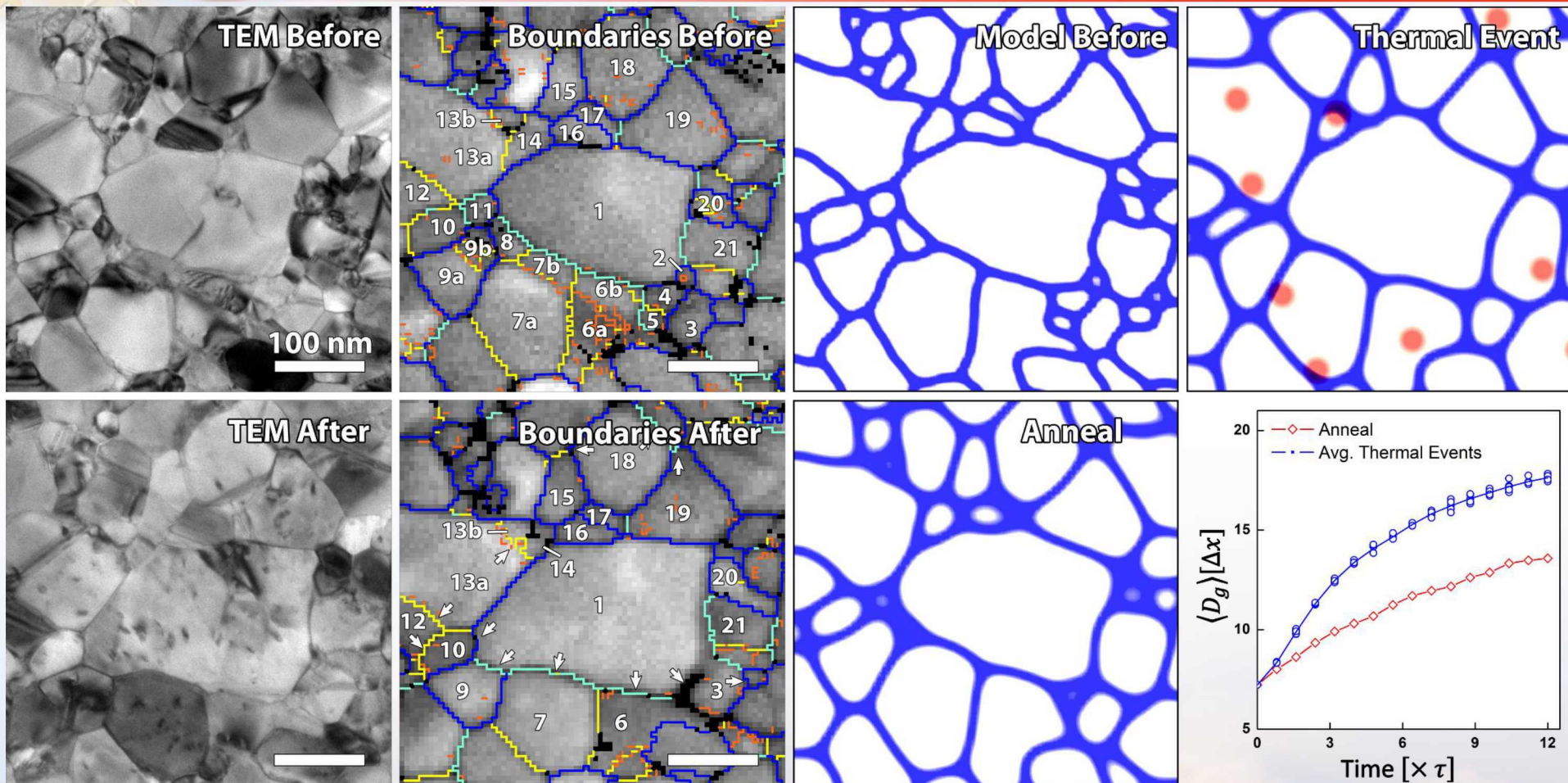
10 MeV Si

Any texture or grain boundary evolution can be directly observed and quantified



# Direct Comparison to Mesoscale Modeling

Collaborators: D.C. Bufford, F. Abdeljawad, & S.M. Foiles

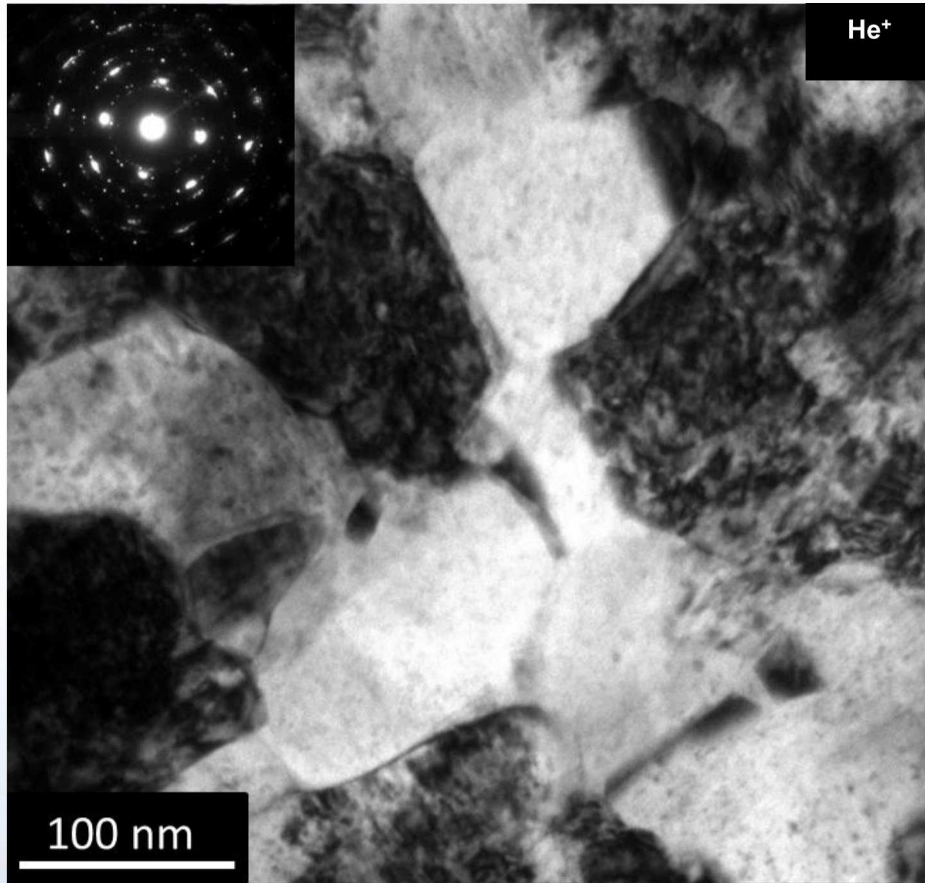


Because of the matching length scale, the initial microstructure can serve as direct input to either MD or mesoscale models & subsequent structural evolution can be directly compared.



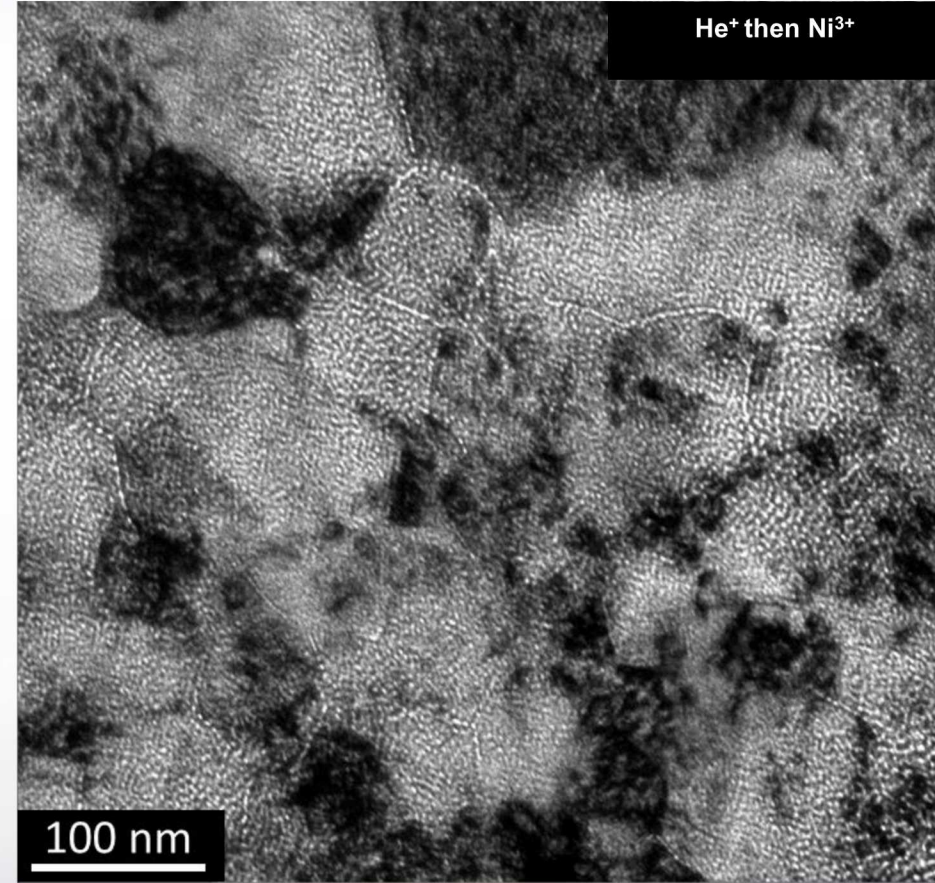
# Heterogeneous Bubble Formation under Some Radiation Sequences

Collaborator: B. Muntifering & J. Qu



$10^{17} \text{ He}^+/\text{cm}^2$

Visible damage to the sample



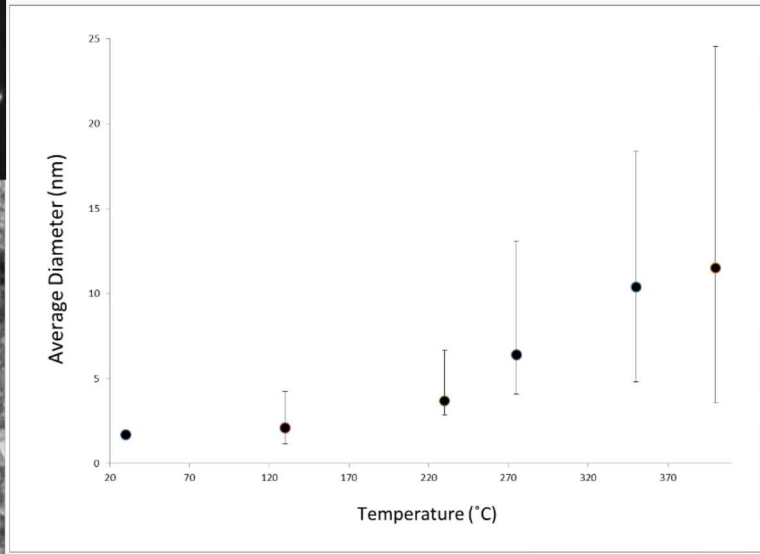
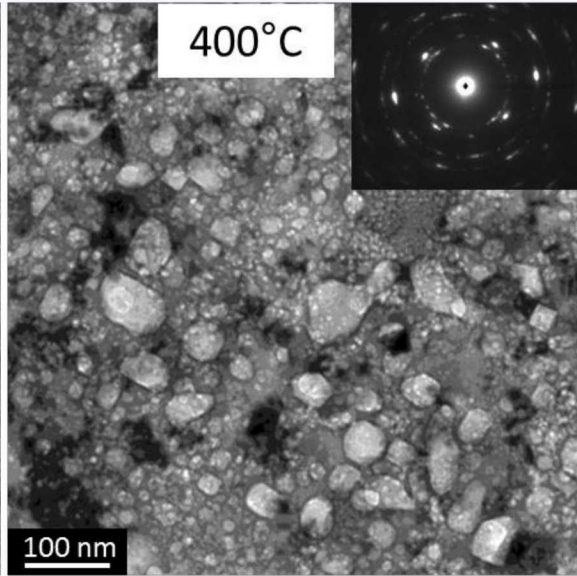
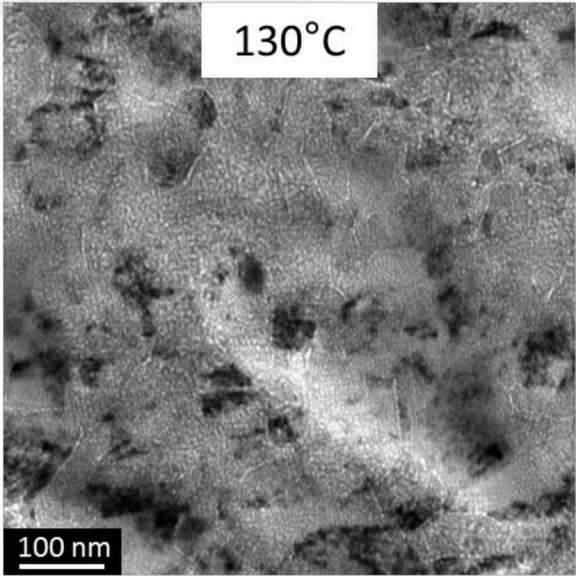
$0.7 \text{ dpa Ni}^{3+}$  irradiation

High concentration of cavities along grain boundaries

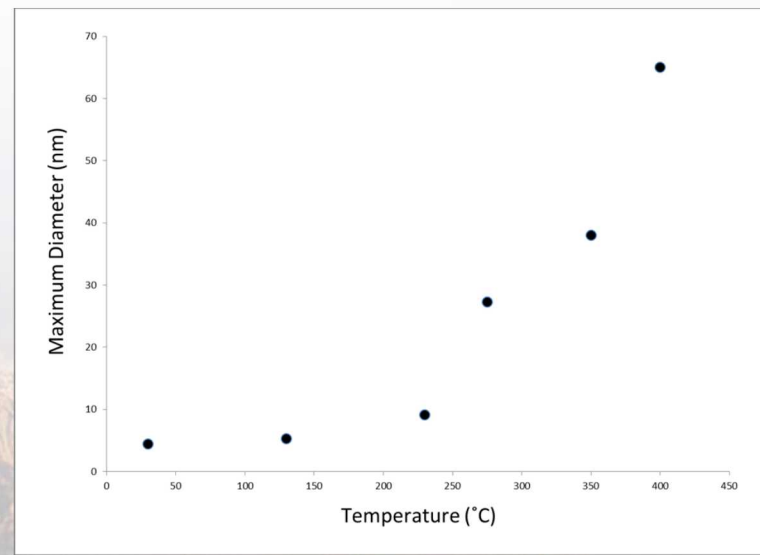
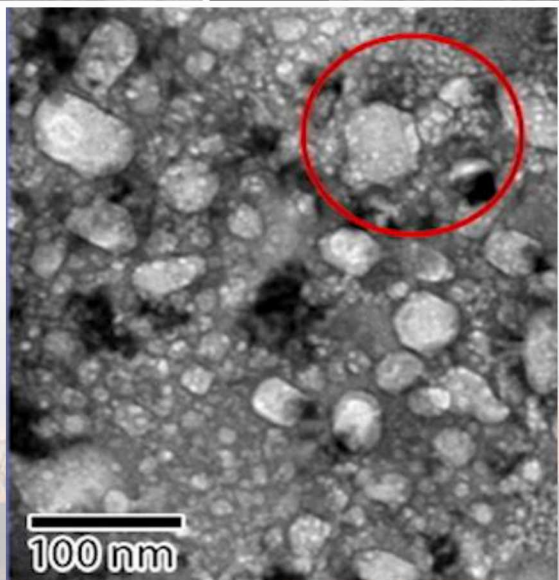


# Cavity Growth during *In situ* Annealing of 10 keV He<sup>+</sup> Implanted and then 3 MeV Irradiated Ni<sup>3+</sup>

Collaborator: B. Muntiferog & J. Qu



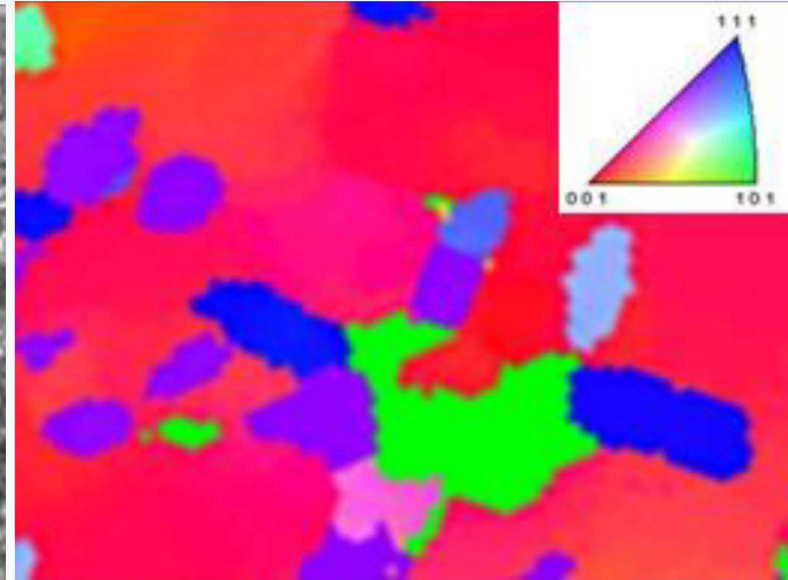
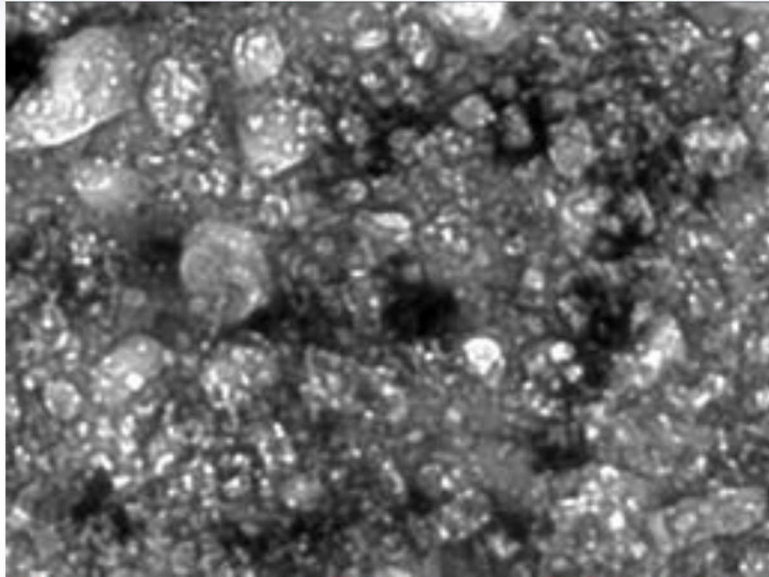
Bubble to cavity transition and cavity evolution can be directly studied



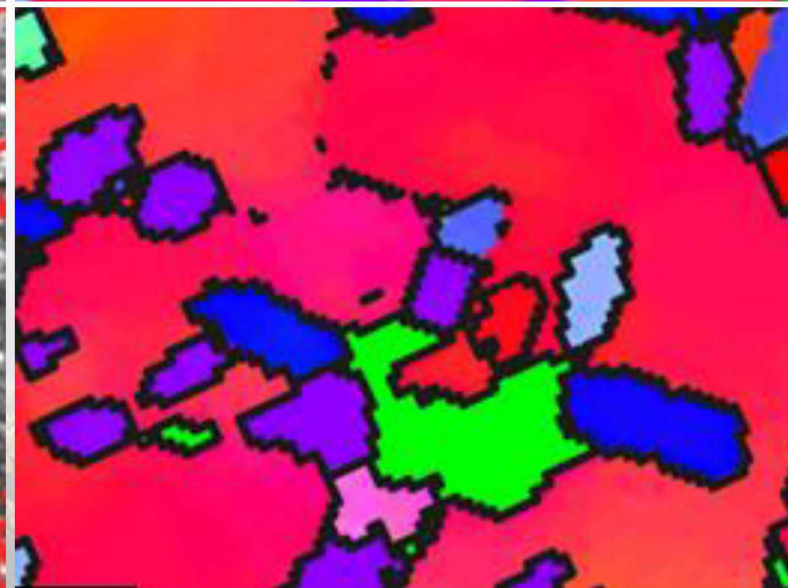
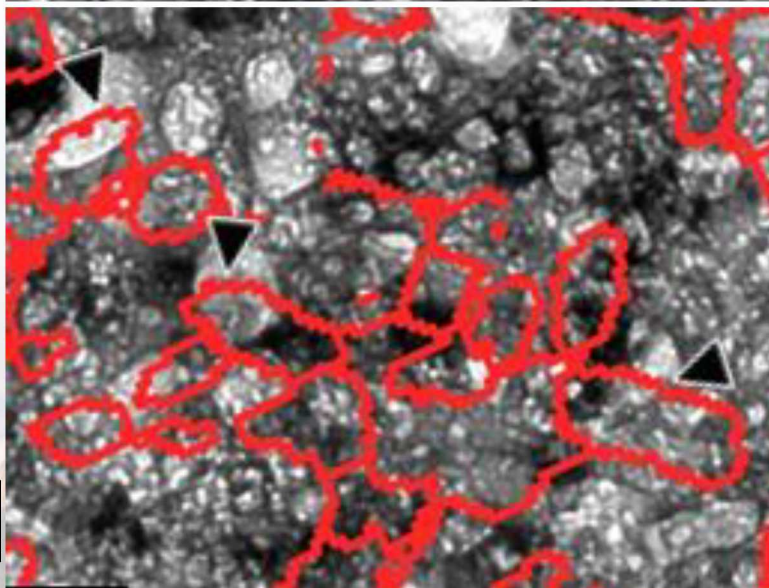
# Precession Electron Diffraction Reveals Hidden Grain Structure

Collaborator: B. Muntifering & J. Qu

Cavities in helium implanted, self-ion irradiated, nc nickel film annealed to 400 °C



Cavities span multiple grains at identified grain boundaries



100 nm

# Bubble Migration and Growth in a Single Grain Boundary

Collaborator: C. Taylor, B. Muntifering, J. Sugar & D. Adams

573 K

50 nm

623 K

50 nm

673 K

50 nm

723 K

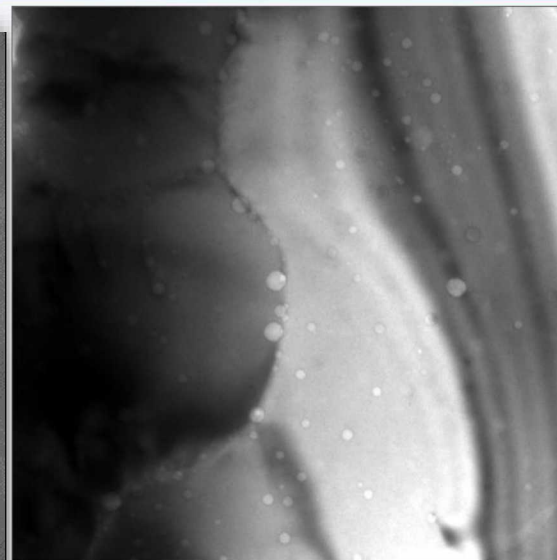
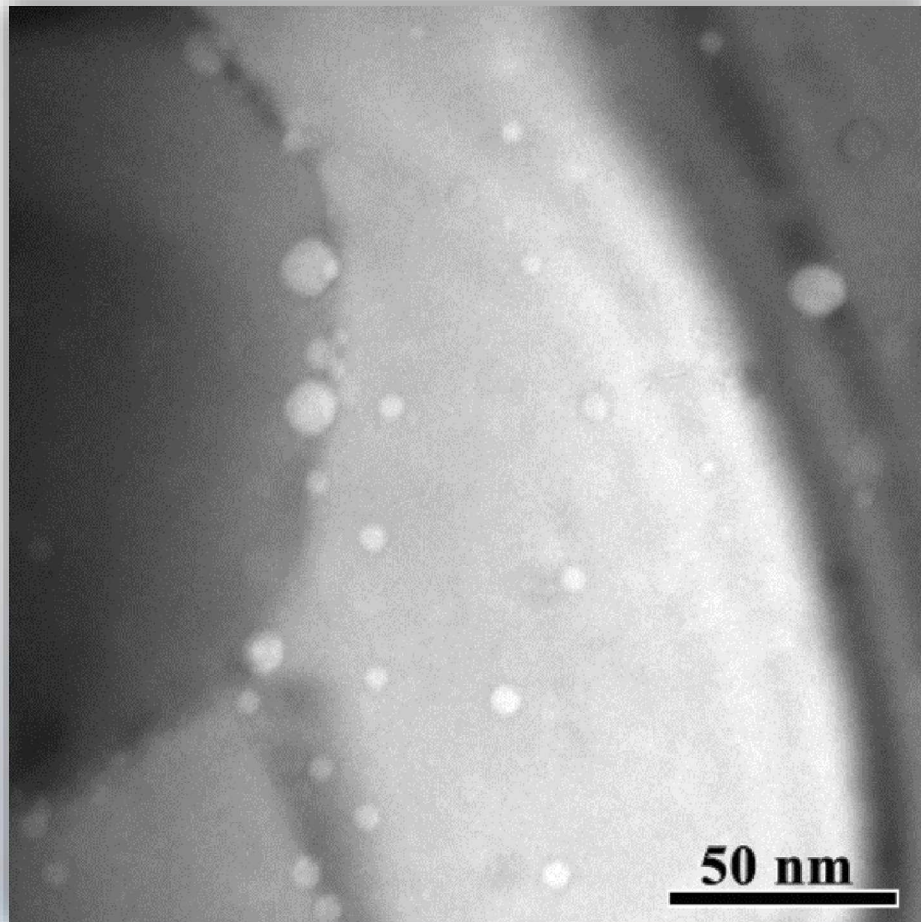
50 nm

- Growth at GB from diffusion of interstitial He or small He-V clusters
- Growth at GB from diffusion of interstitial He or small He-V clusters
- Bubble migration to GB & cavity growth, intragranular growth from diffusion of interstitial He or small He-V clusters
- Blister formation at boundaries, intragranular faceted cavity growth. Growth by cavity diffusion and coalescence.

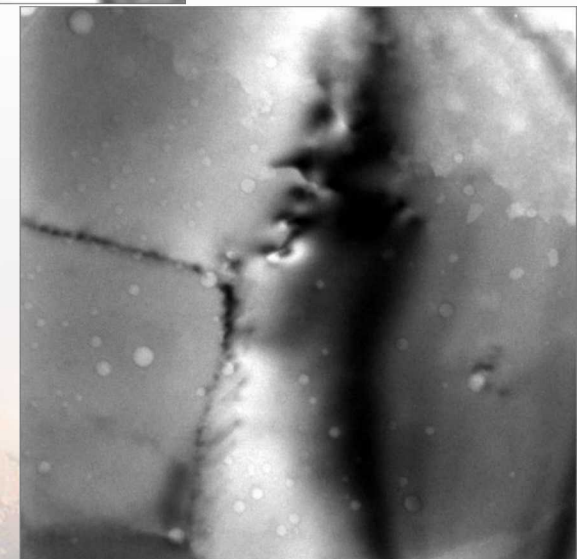


# Cavities Role on Grain Boundary Motion

Collaborator: C. Taylor, B. Muntifering, J. Sugar & D. Adams



Cavities in helium implanted, Pd foil during annealing at 700 °C



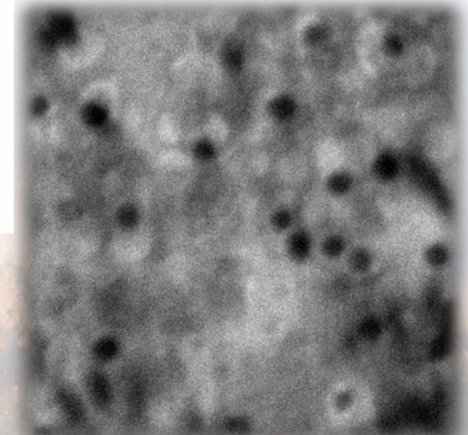
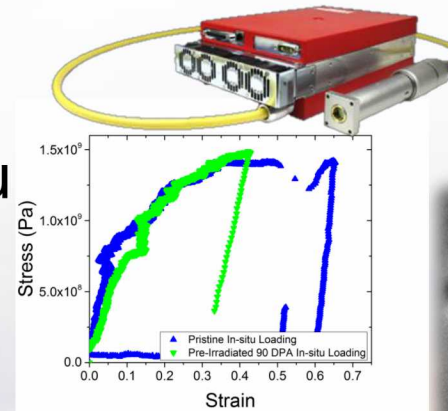
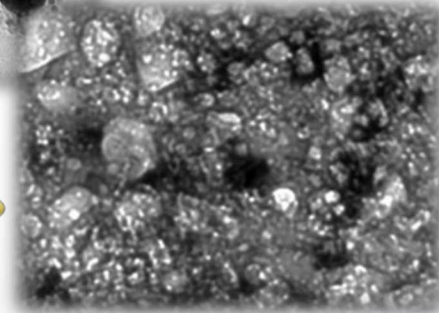
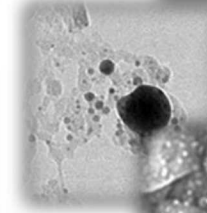
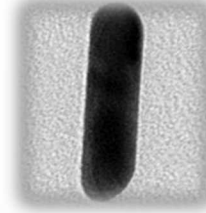
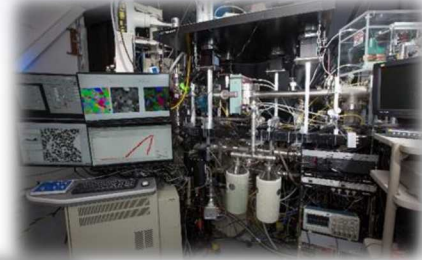
Cavities effect grain boundary mobility, triple junction angle



grain boundary motion alters cavity coalescence

# Outline

1. Introduction to Sandia National Laboratories and its *In situ* Ion Radiation Transmission Electron Microscope (I<sup>3</sup>TEM)
2. Understanding radiation stability in nanoparticles
3. Effect of radiation environments on nanocrystalline metals
4. Lasers in a TEM (why not?)
5. State-of-art in quantitative in situ mechanical testing
6. Other environments (in situ SEM, liquid, and gas)



# Initial Laser Heating Observations

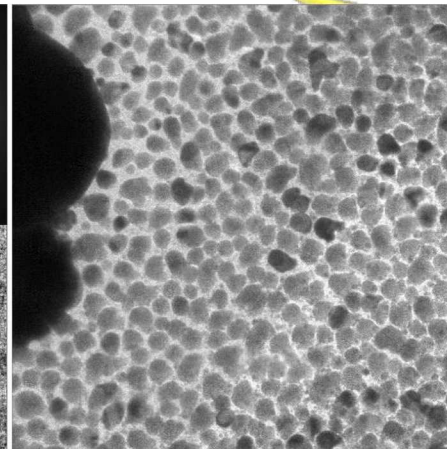
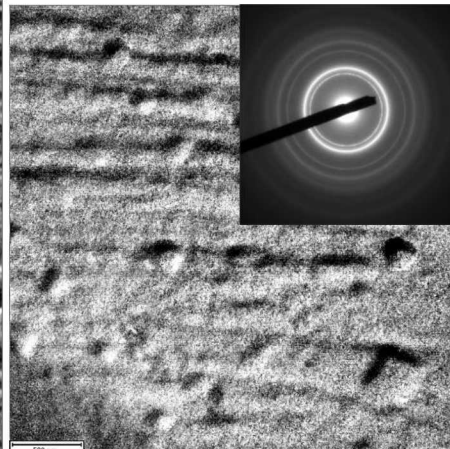
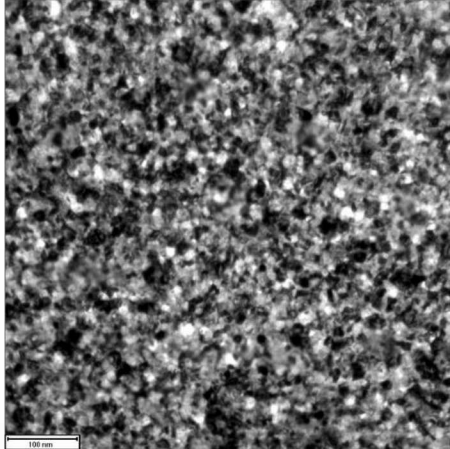
Collaborator: P. Price, C.M. Barr, D. Adams, M. Abere

Pt Grain Growth

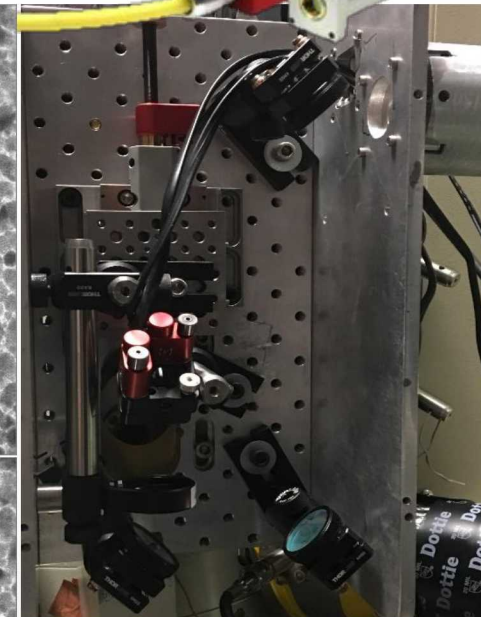
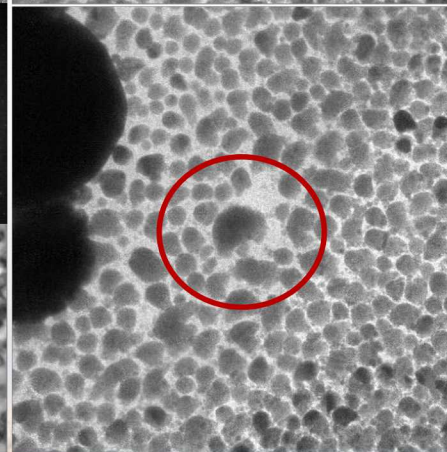
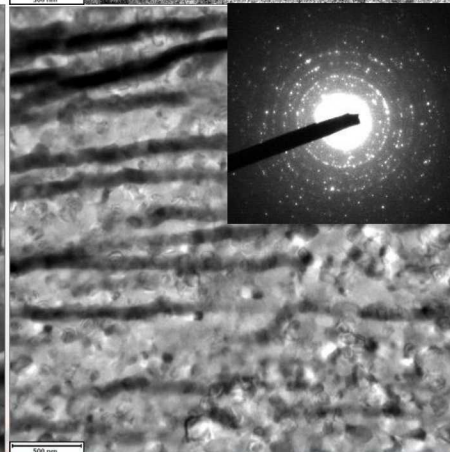
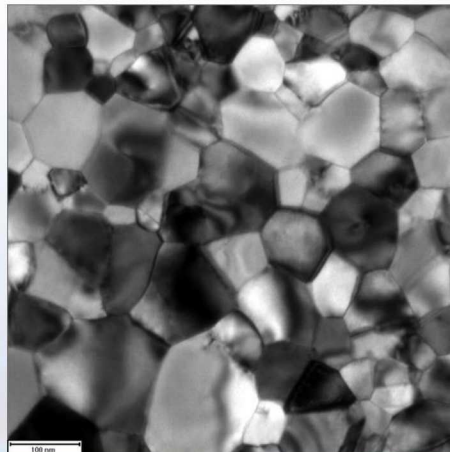
Reactive Multilayer Films

Nanoparticle Sintering

Before



After



Electron Beam

IR Laser

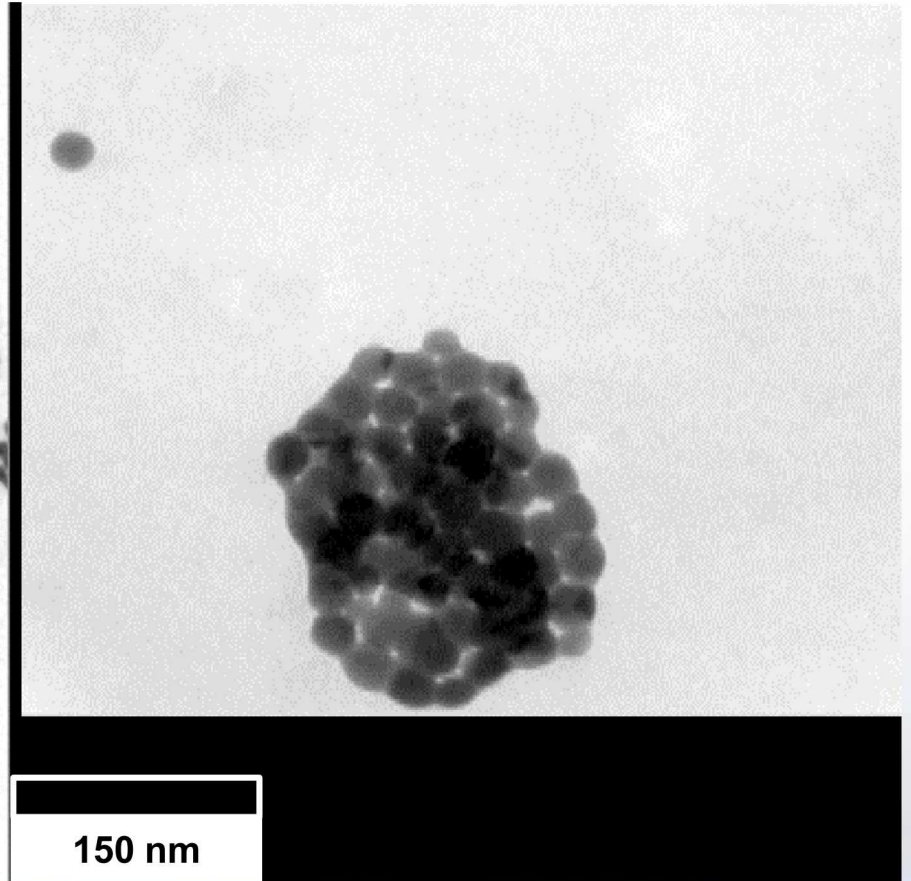
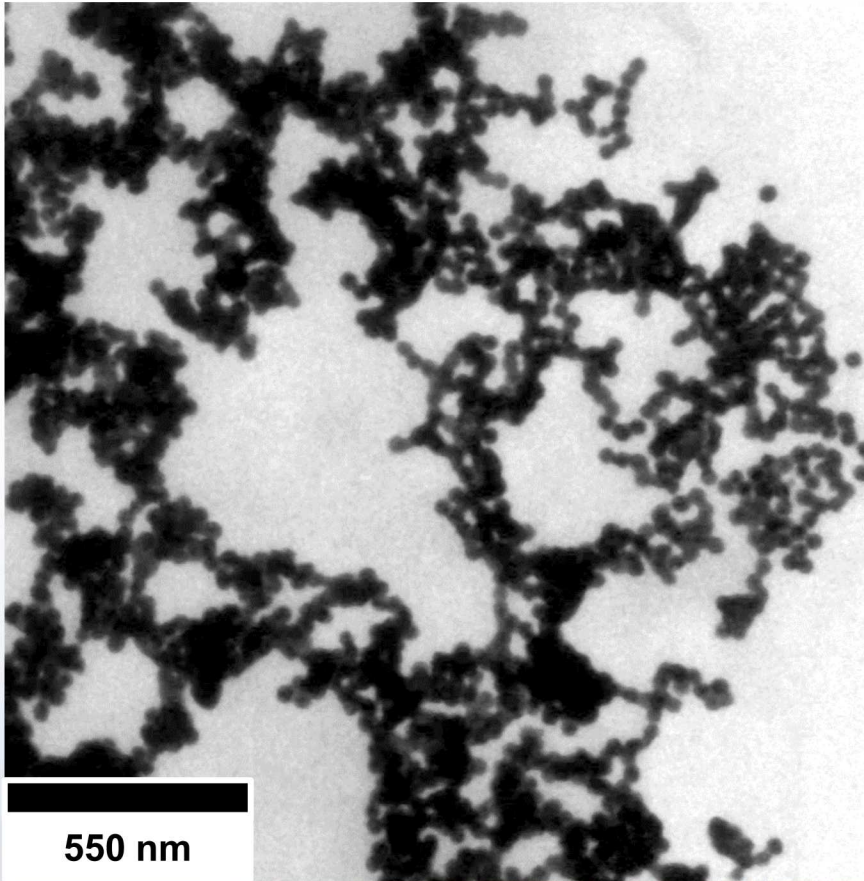
We can now introduce rapid thermal heating with any TEM stage or ion beam conditions



Sandia National Laboratories

# Complex Interaction Au NPs Exposed to Laser Irradiation

Contributors: P. Price, L. Treadwell, A. Cook



Speed = 2.5x



**A Complex Combination of Sintering, Reactions, and Ablation Occurs**

# $\mu\text{s}$ Resolution with a Standard Camera

Collaborator: P. Price, A. Monterrosa, D. Adams, M. Abere, & IDES Inc.

fs

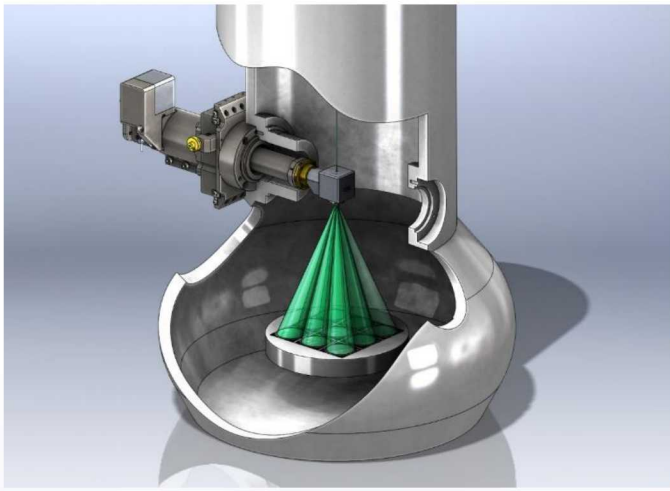
ps

ns

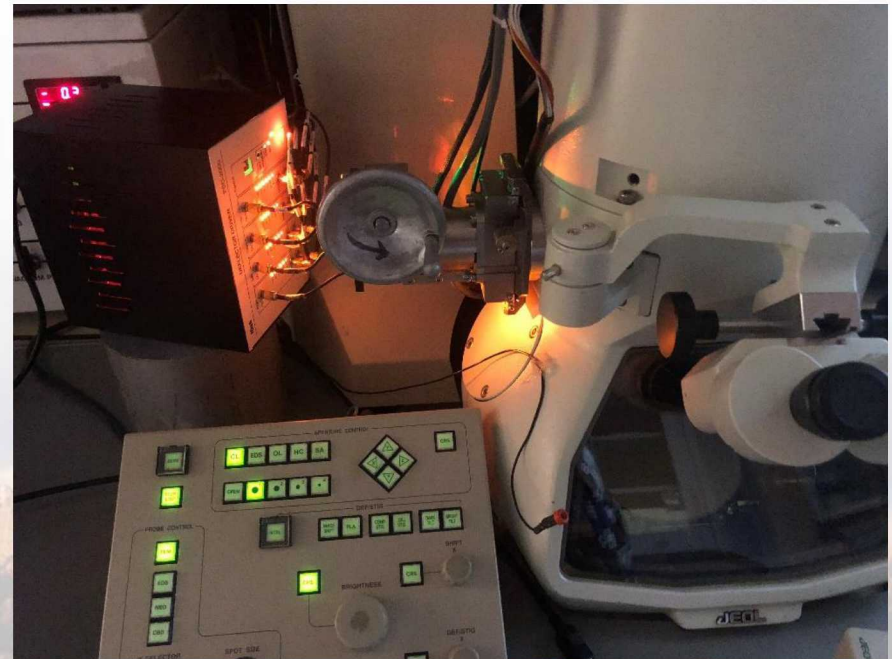
$\mu\text{s}$

ms

s

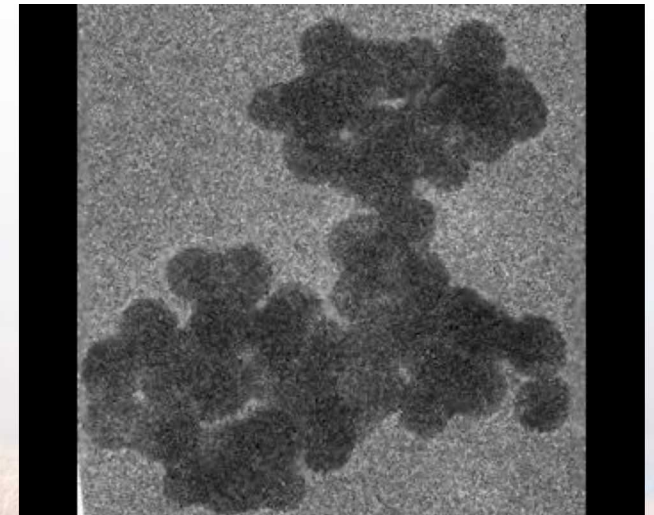
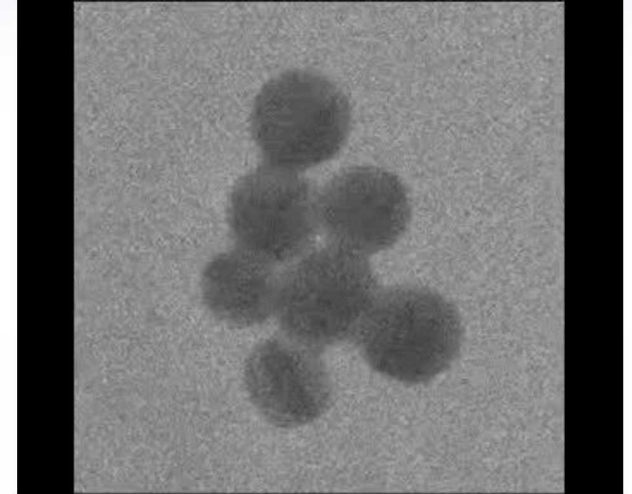
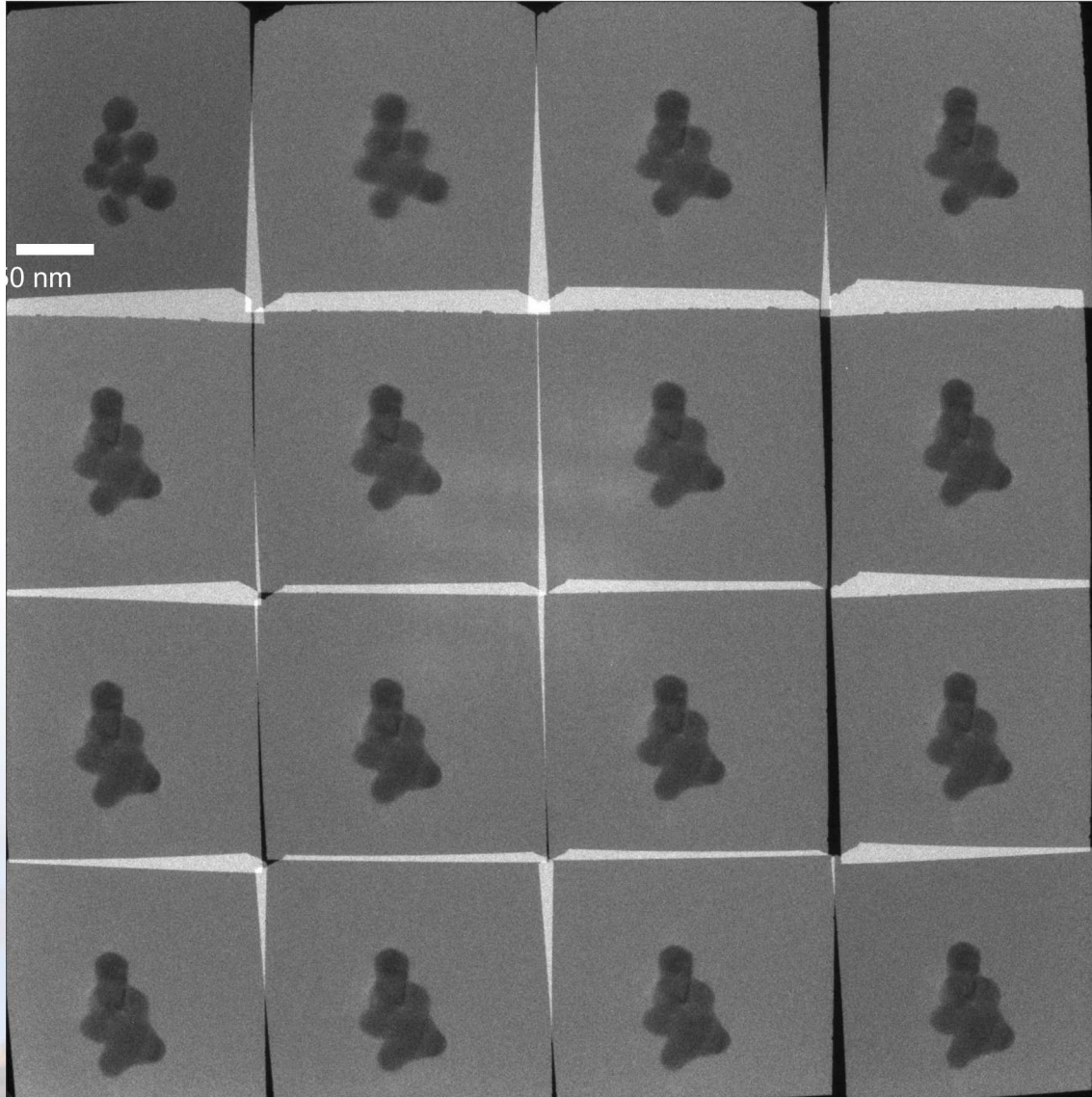


- Electrostatic deflection of electrons
- 4, 9, or 16 images per frame, spread over a large camera
- Any exposure time up to the limits of the camera
  - Ultimate limit is beam current/brightness



# 1-to-1 Frame Capture (<5 ms per frame) Sintering of 20 nm Au Nanoparticles

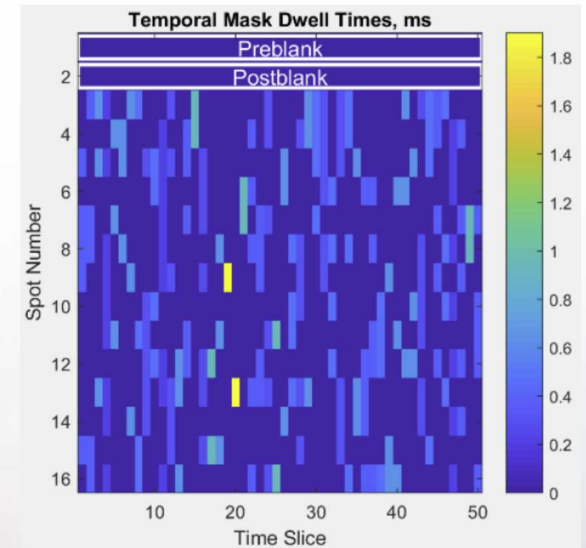
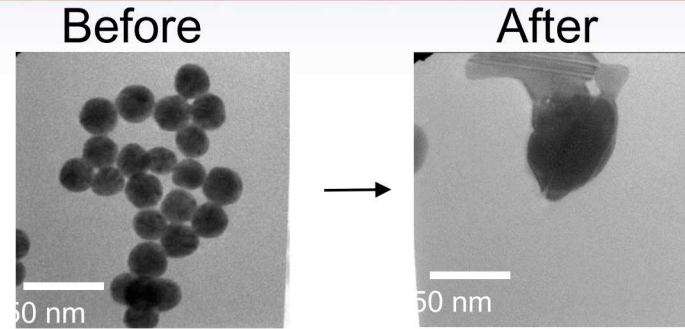
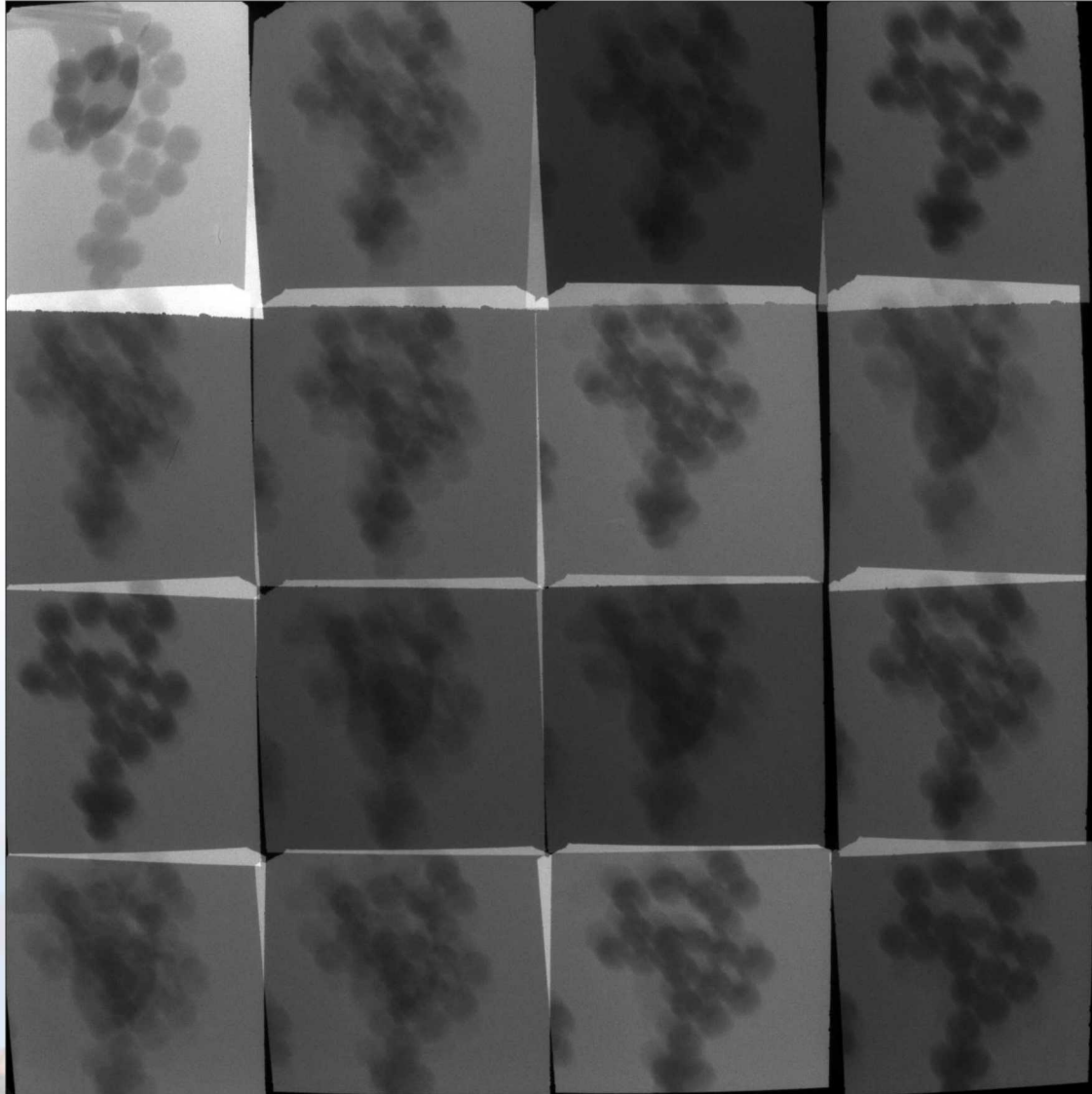
Collaborator: P. Price, A. Monterrosa, & IDES Inc.



16 frames captured with <5 ms exposure per frame

# Temporal Compressive Sensing to Improve Temporal Resolution

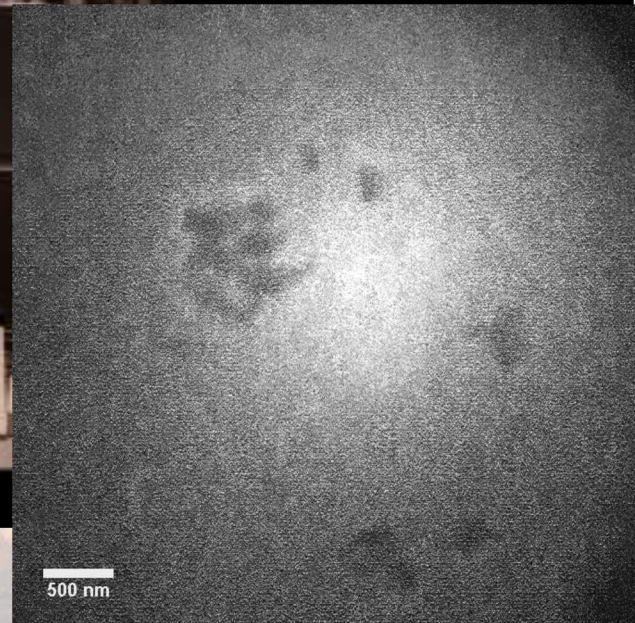
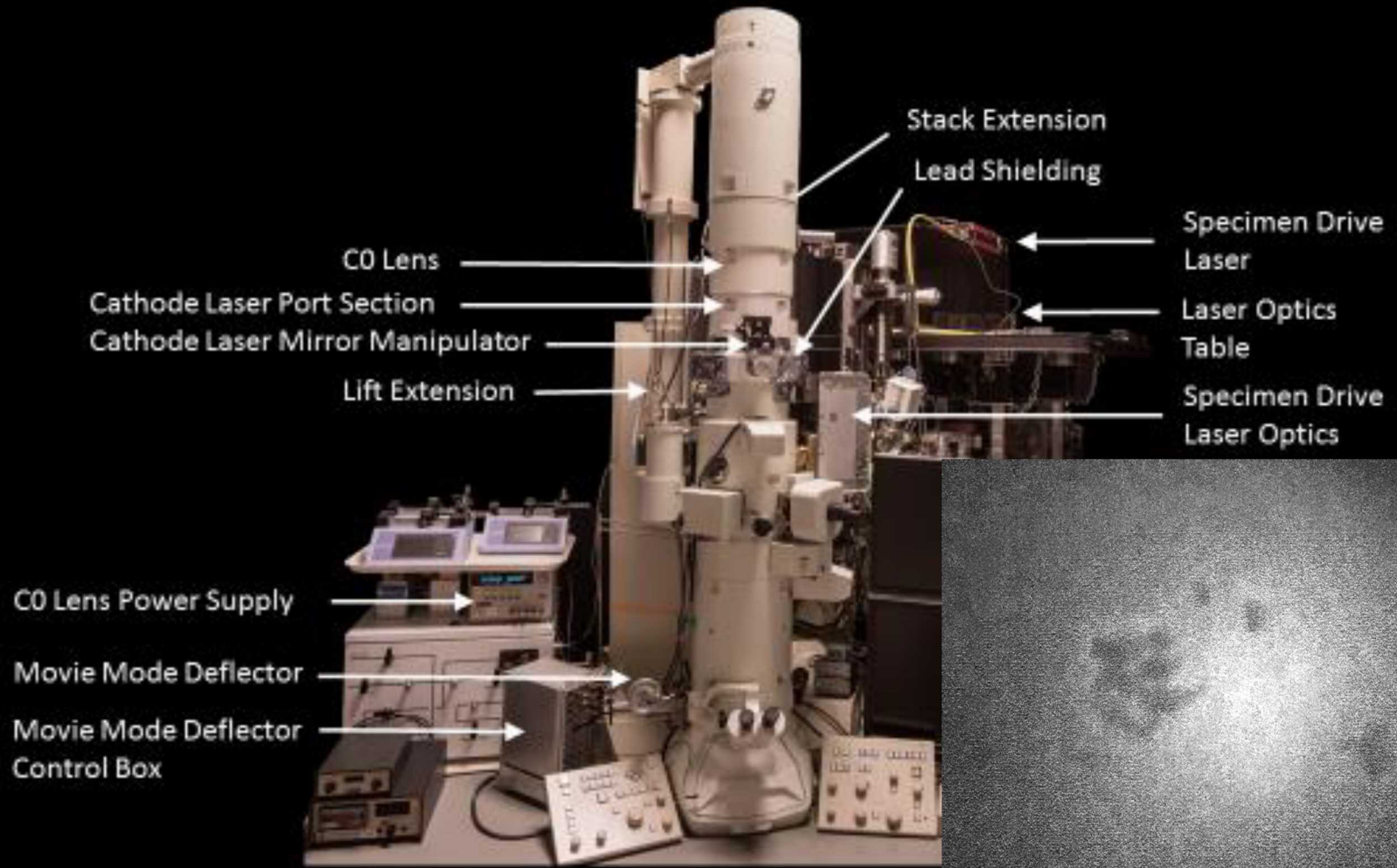
Collaborator: P. Price, A. Monterrosa, & IDES Inc.



**A pseudorandom exposure pattern can produce more than 16 frames within the same exposure time**

# Current Status of DTEM Conversion

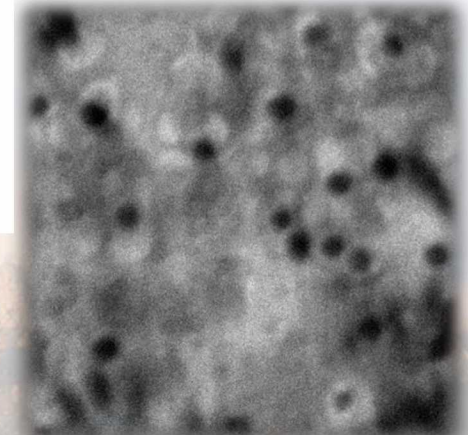
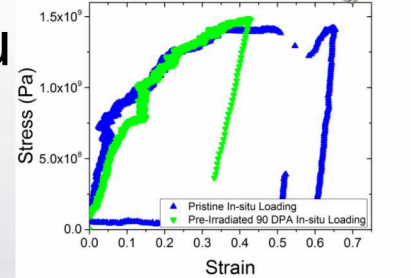
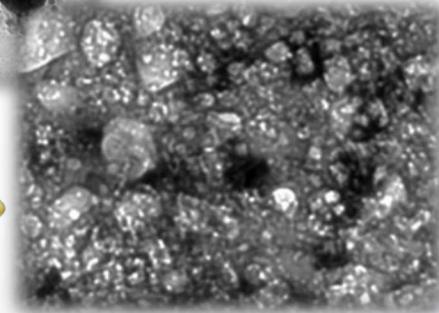
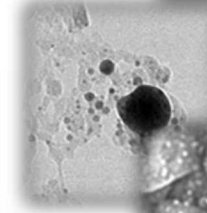
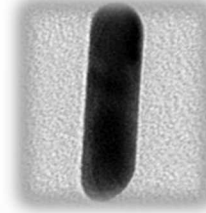
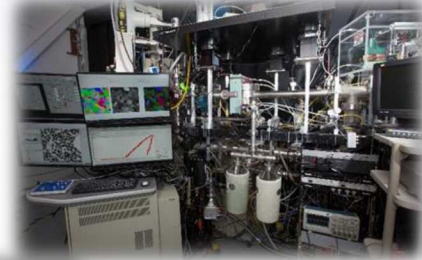
Collaborator: P. Price, A. Monterrosa, C.M. Barr, D. Adams, M. Abere, & IDES Inc.



- 266 nm UV laser induced photoemission has been achieved!
- 6 ns single-shot DTEM image of P47

# Outline

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6. Other environments (in situ SEM, liquid, and gas)



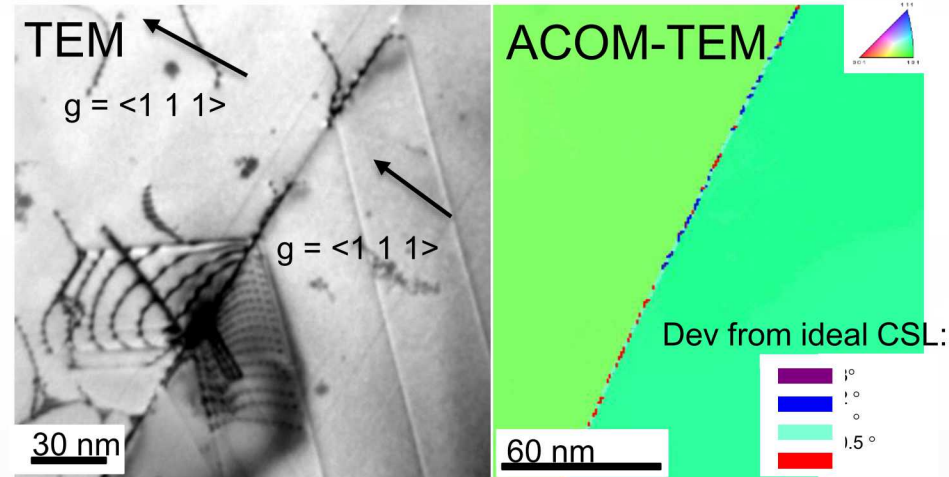
# In situ Qualitative Mechanical Testing

Collaborators: C. Barr

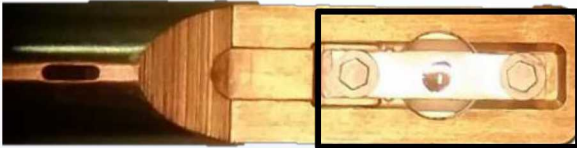
## Gatan straining TEM Holder

- Minimal control over displacement and no “out-of-box” force information
- Successful in studies in observing dislocation-GB interactions/mechanisms
- Ideally both grains have kinematic BF 2-beam conditions: challenging in ST holder

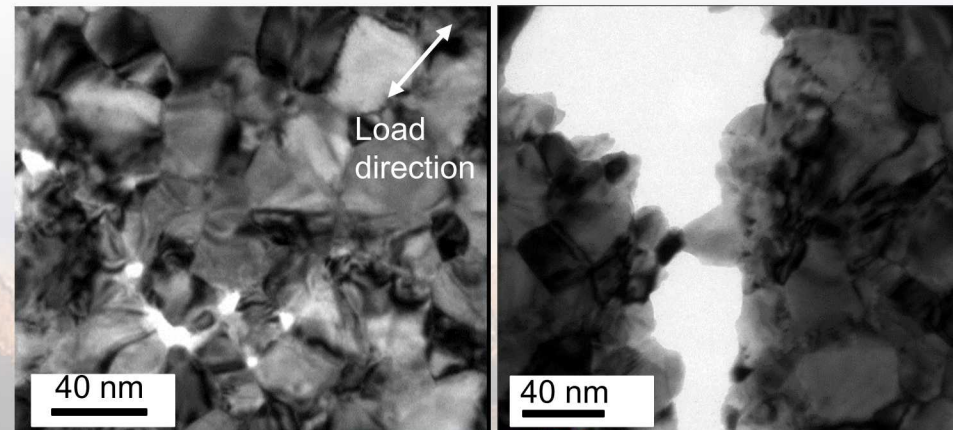
Dislocation interactions as a function of GB character ( $\Sigma 3$  twin GB below):



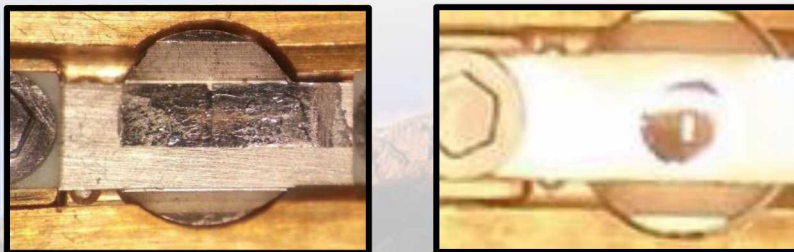
## Traditional Gatan Heating and Straining Holder



Observe deformation mechanisms in nanocrystalline metals during tensile straining:



Thin film tension “jig”: Jet thinned disk:



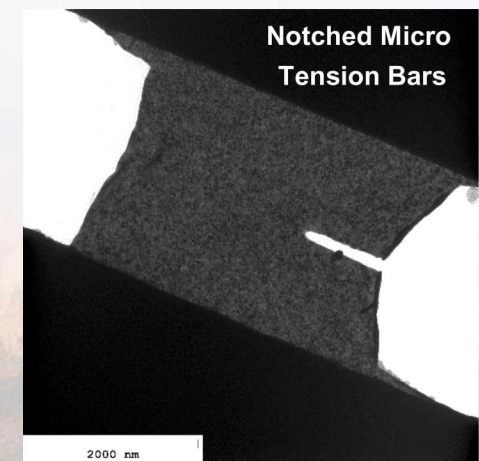
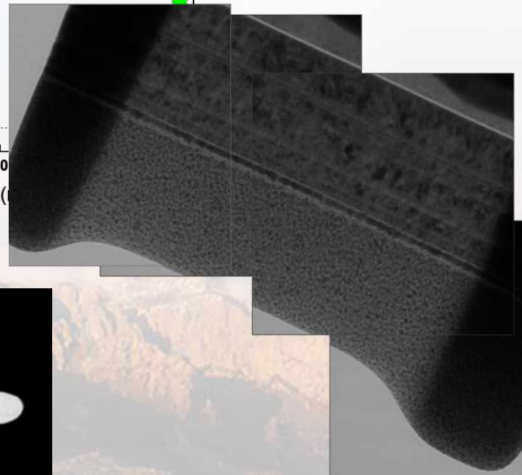
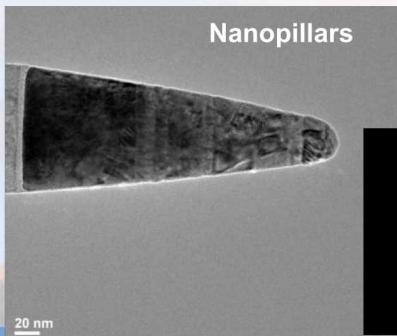
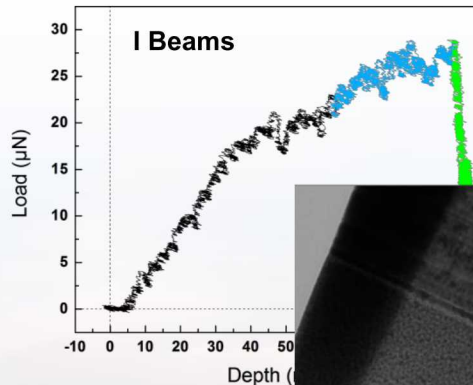
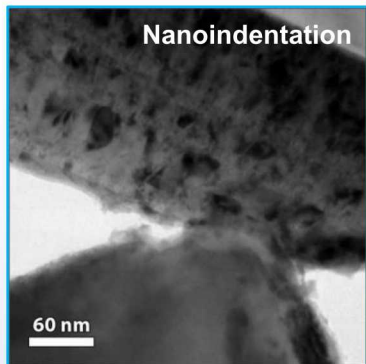
# *In situ* Quantitative Mechanical Testing

Contributors: J. Sharon, B. L. Boyce, C. Chisholm, H. Bei, E.P. George, P. Hosemann, A.M. Minor, & Hysitron Inc.



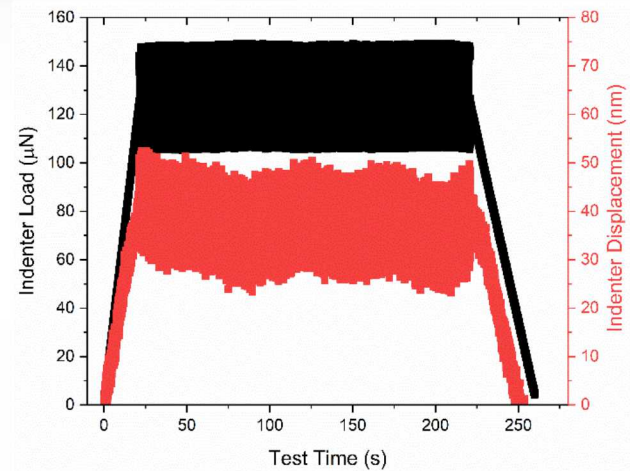
Hysitron PI95 *In Situ* Nanoindentation TEM Holder

- Sub nanometer displacement resolution
- Quantitative force information with  $\mu\text{N}$  resolution
- **Concurrent real-time imaging by TEM**



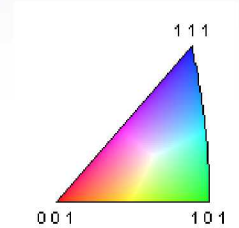
# Cyclic Loading Coupled with ACOM

Collaborators: C. Barr & W. Mook

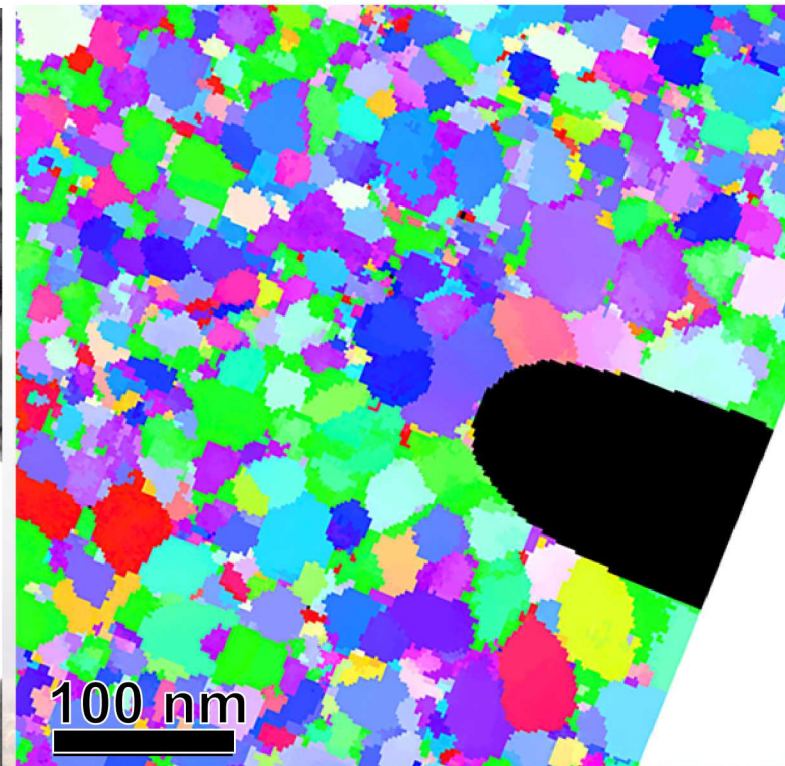
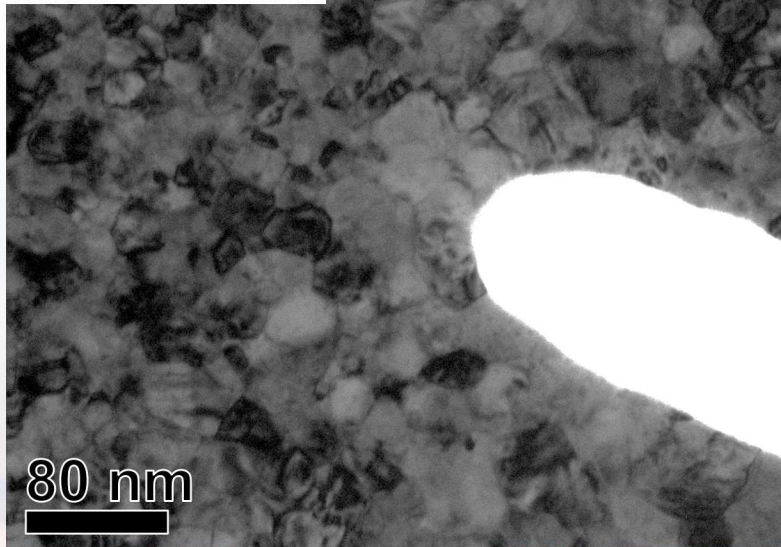


Mean load ( $P_{\text{mean}}$ ) = 135  $\mu\text{N}$

Amplitude load ( $P_{\text{amp}}$ ) = 35  $\mu\text{N}$



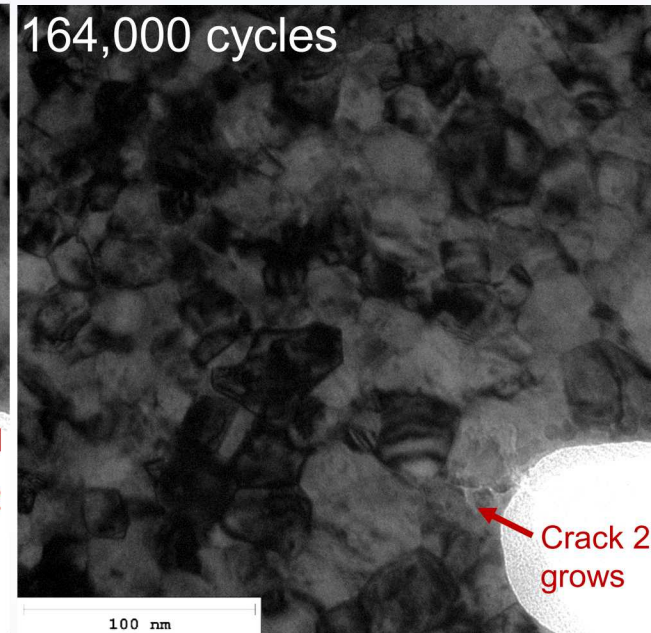
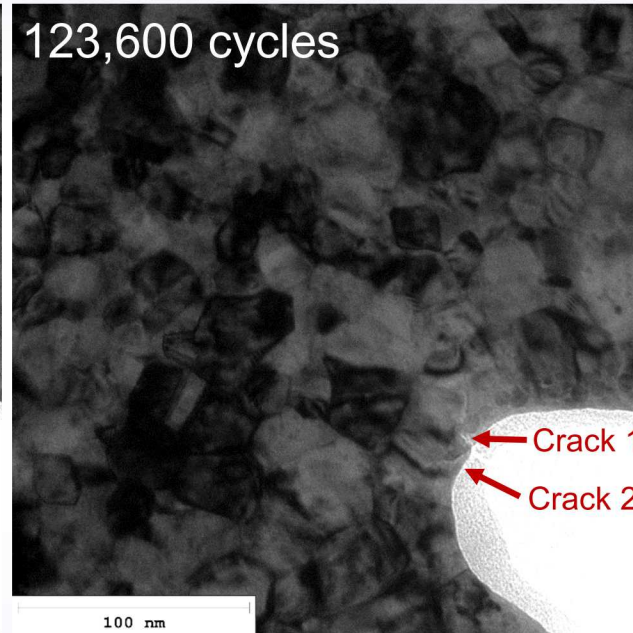
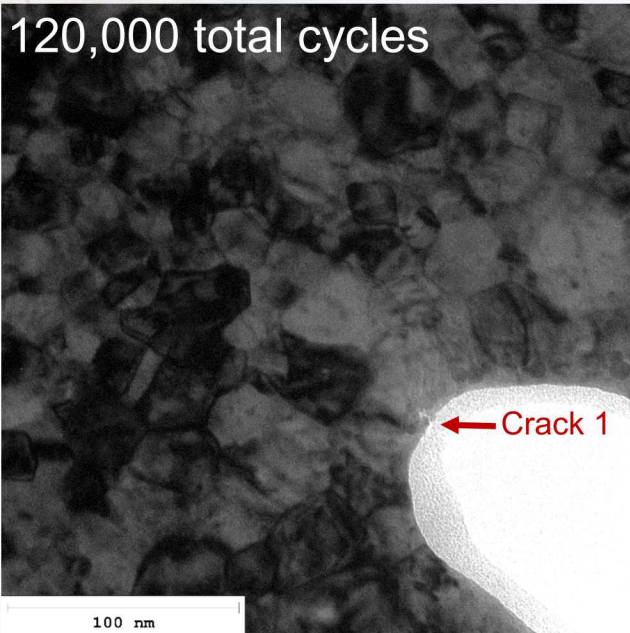
Loading direction



Orientation maps pre-, intermediate, and post- in-situ mechanical test can assist in deconvoluting possible mechanisms during cyclic loading

# Crack Initiation at Notch

Collaborators: C. Barr & W. Mook

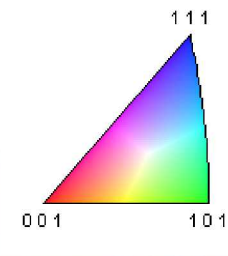
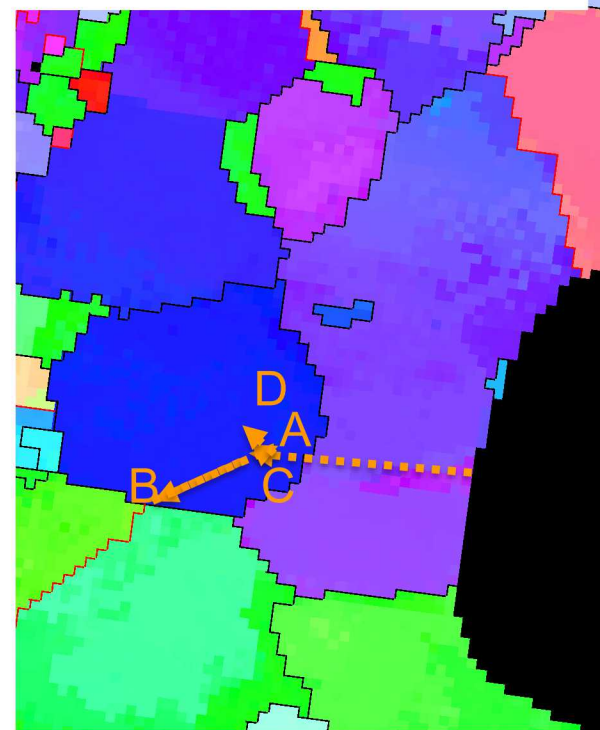
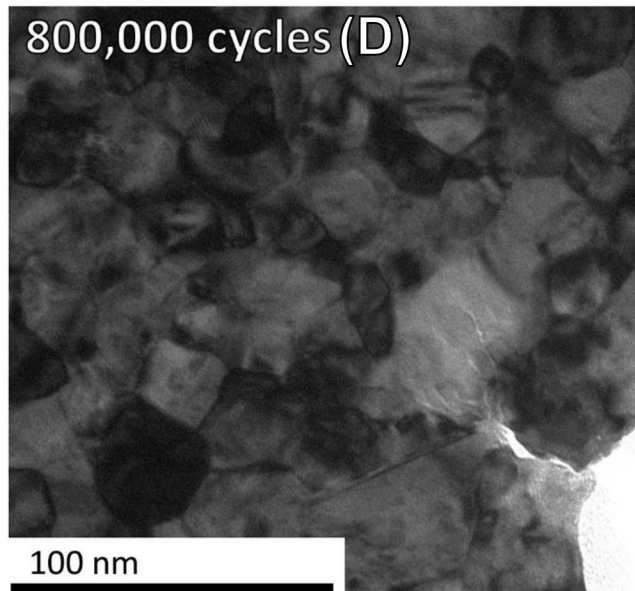
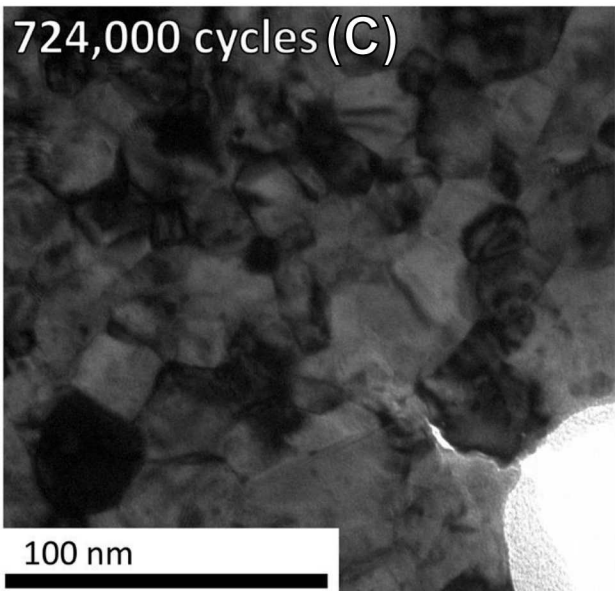
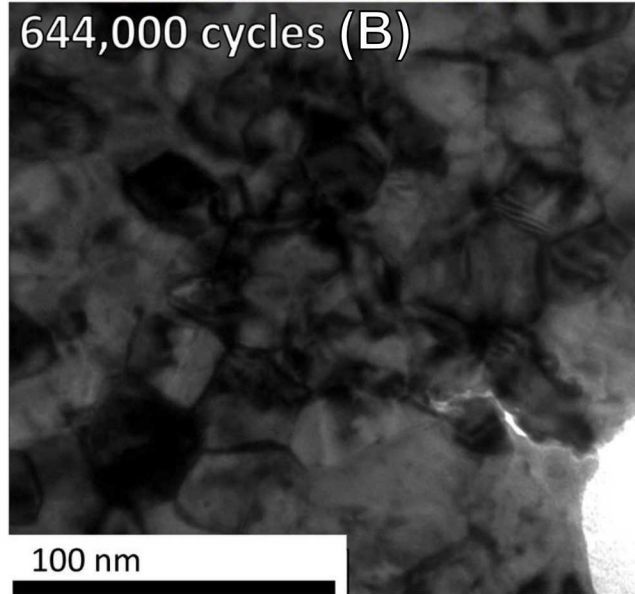
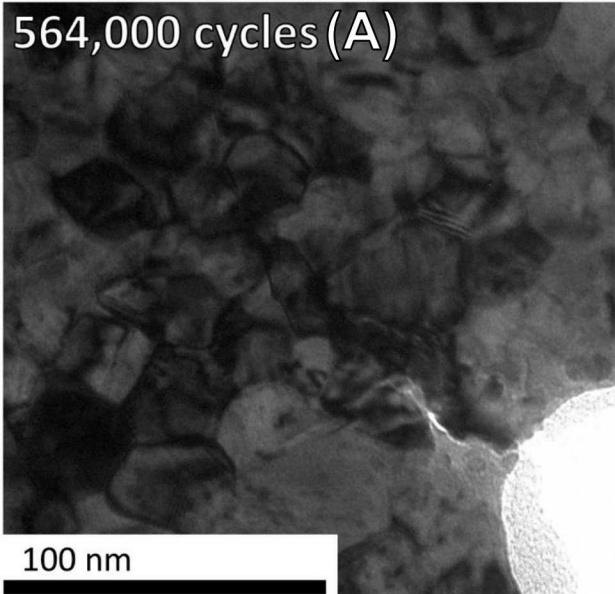


- Crack initiation and initial propagation at notch tip
- Second crack initiates at  $\sim 90^\circ$  to first crack, both  $45^\circ$  to notch tip normal
- Intra-granular crack (crack #2) propagates until reaching initial grain boundary and is subsequently arrested



# Crack Propagation, Closure, and Re-Direction

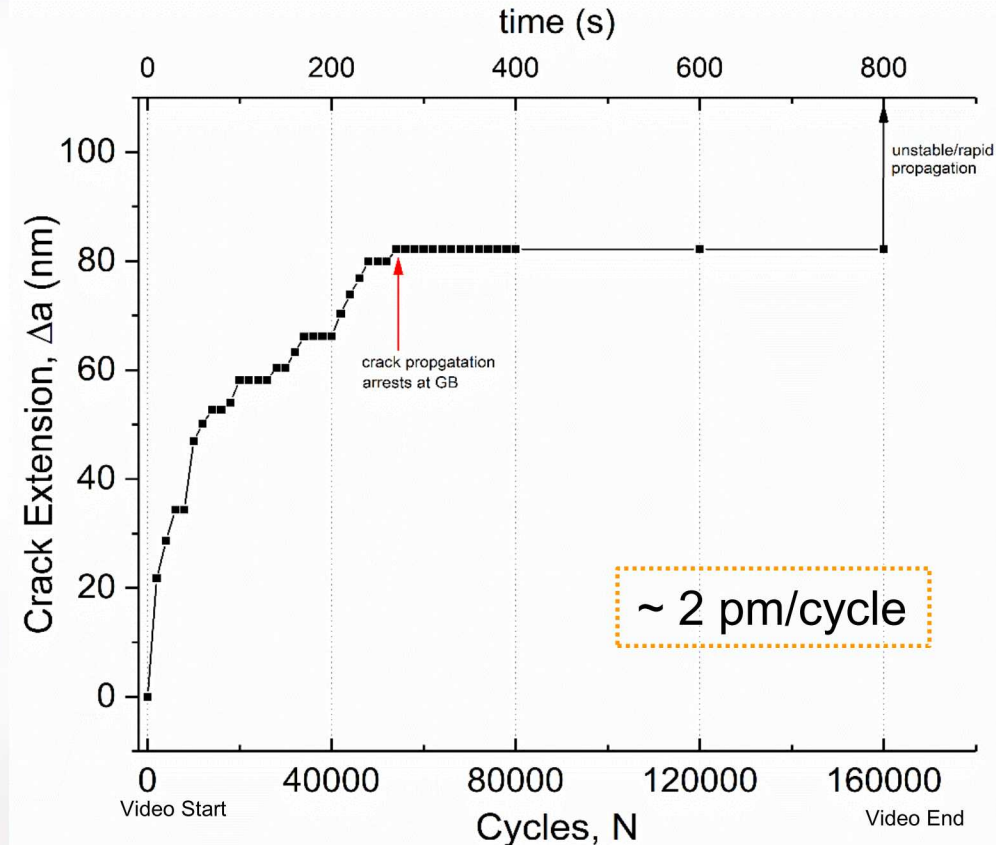
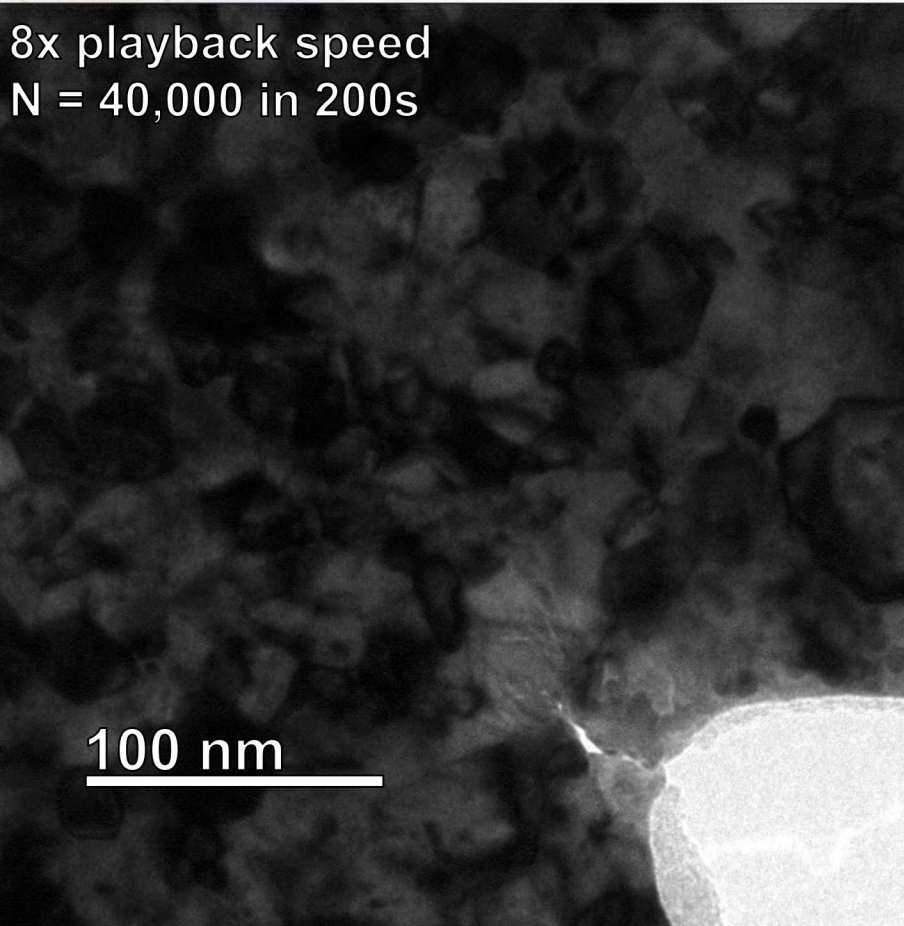
Collaborators: C. Barr & W. Mook



# Cyclic Loading: Complex Crack Propagation

Collaborators: C. Barr & W. Mook

8x playback speed  
N = 40,000 in 200s



- Mean load: 135  $\mu$ N; Amplitude load: 35  $\mu$ N
- 200 Hz, 200s test (15 fps 1k x 1k camera)

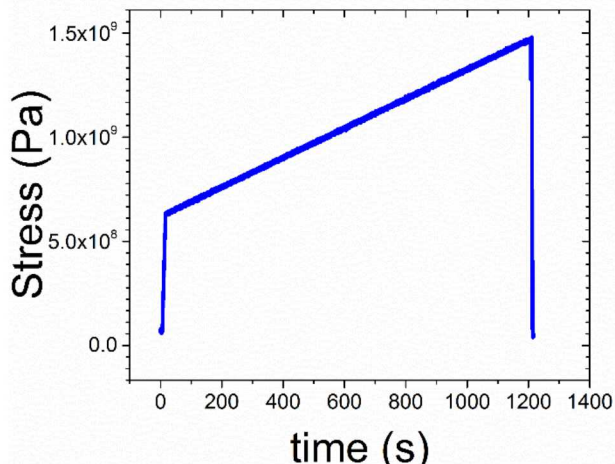
- $da/dN = 1.7 \times 10^{12}$  m/cycle
- Non-linear crack extension rate
- Crack propagation path changes "direction"



# Irradiation Creep (4 MeV Cu<sup>3+</sup> 10<sup>-2</sup> DPA/s)

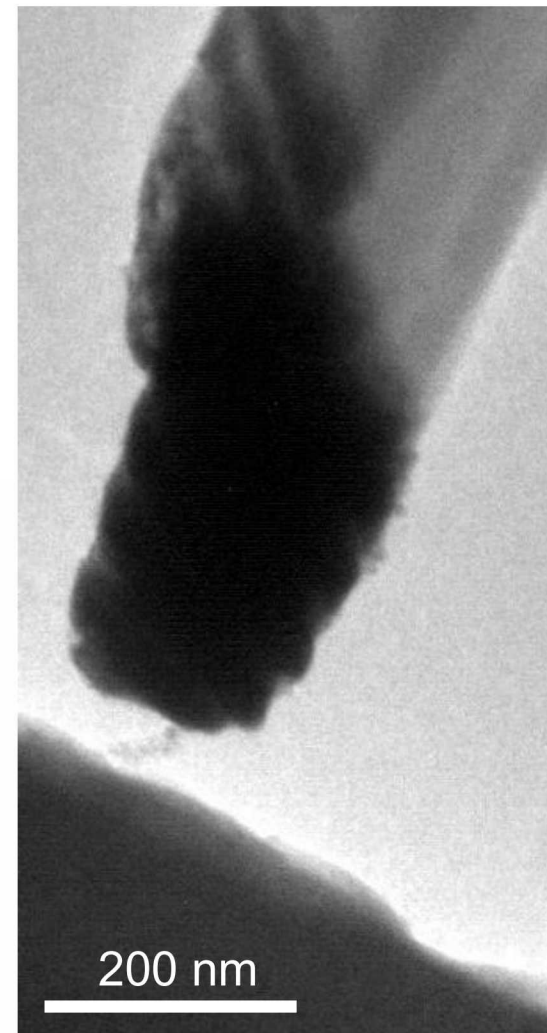
Contributors: G.S. Jawaharram, S. Dillon & R.S. Averback

## Controlled Loading Rate Experiments



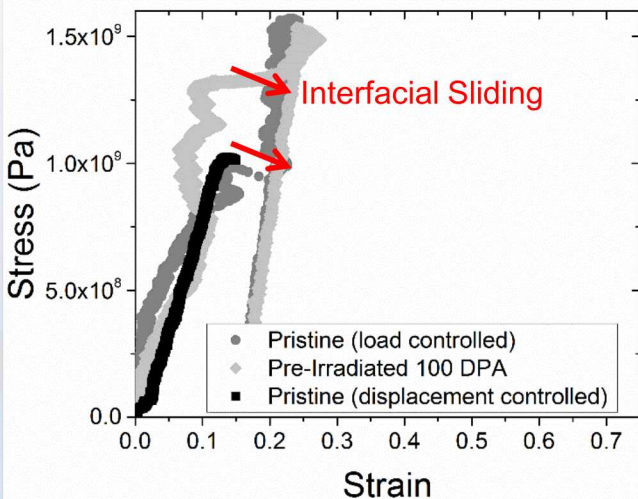
In-situ TEM  
radiation  
creep is  
feasible!

50 nm Cu-W multilayer  
20 Min



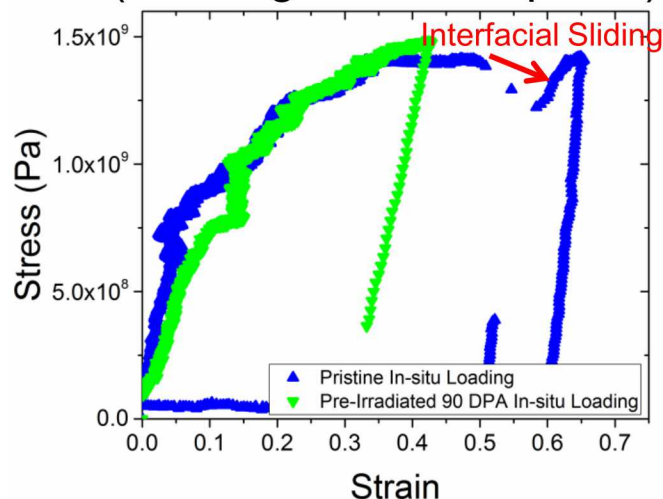
## No Irradiation

(Loading rate 0.6 Mpa s<sup>-1</sup>)



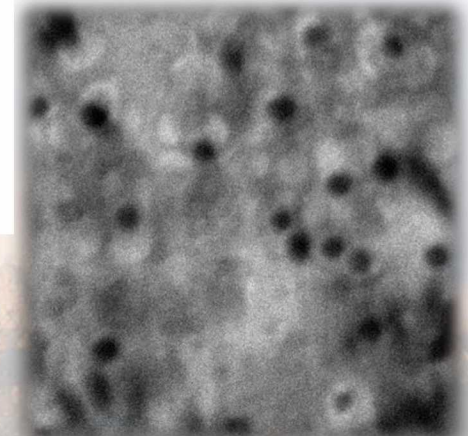
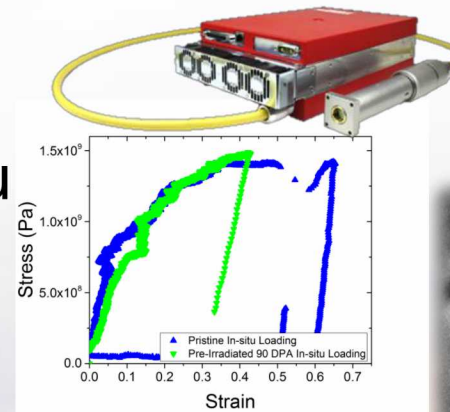
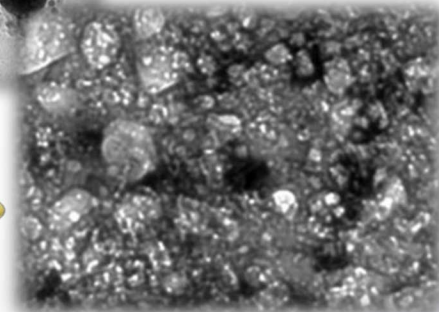
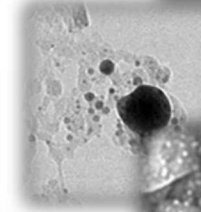
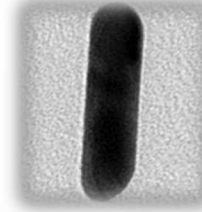
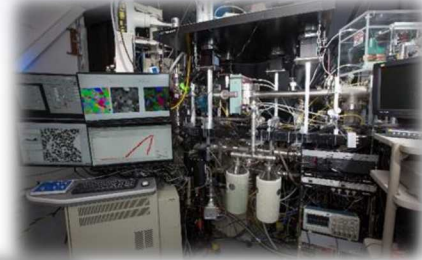
## Irradiation Creep

(Loading rate 0.6 Mpa s<sup>-1</sup>)



# Outline

1. Introduction to Sandia National Laboratories and its *In situ* Ion Radiation Transmission Electron Microscope (I<sup>3</sup>TEM)
2. Understanding radiation stability in nanoparticles
3. Effect of radiation environments on nanocrystalline metals
4. Lasers in a TEM (why not?)
5. State-of-art in quantitative in situ mechanical testing
6. Other environments (in situ SEM, liquid, and gas)

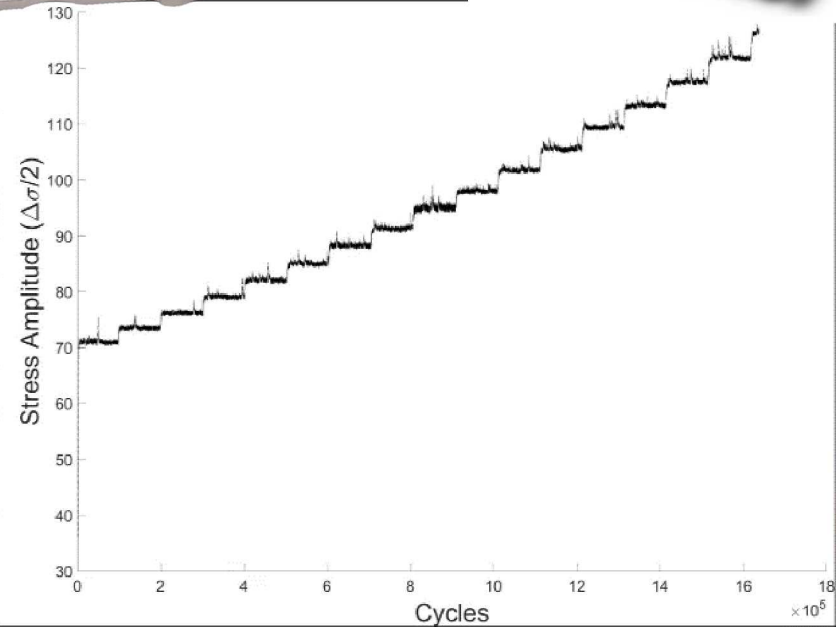
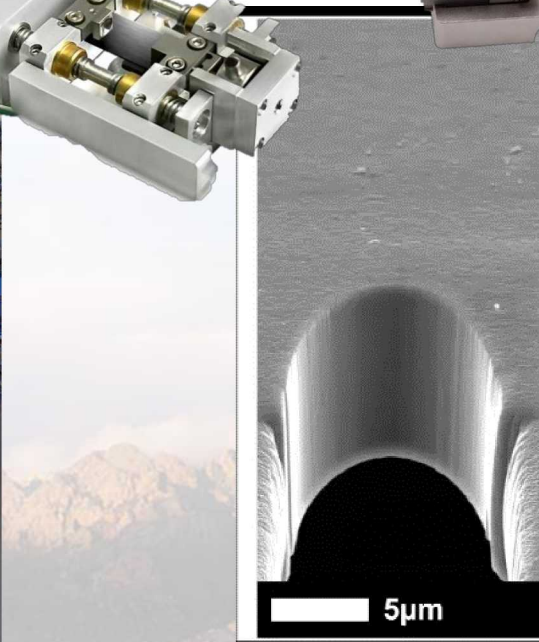
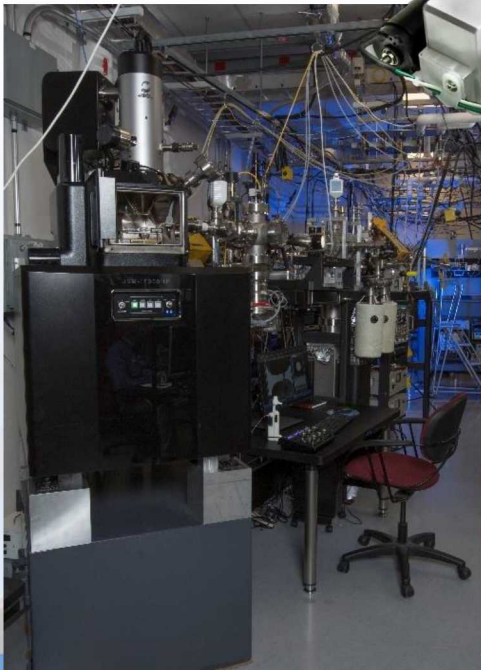
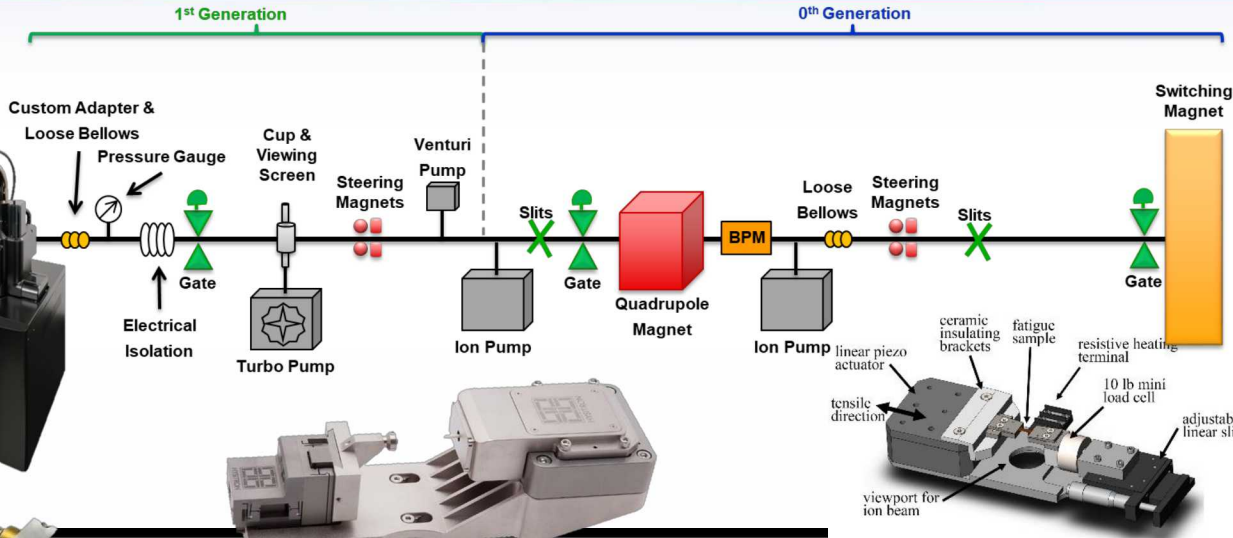


# Scaling Back Up: *In situ* Ion Irradiation SEM (I<sup>3</sup>SEM)

Collaborators: N. Heckman, D. Buller, B. Boyce, J. Carroll, C. Taylor, B. Muntiferung, & S. Briggs



First Beam into SEM  
on April 6<sup>th</sup>, 2018

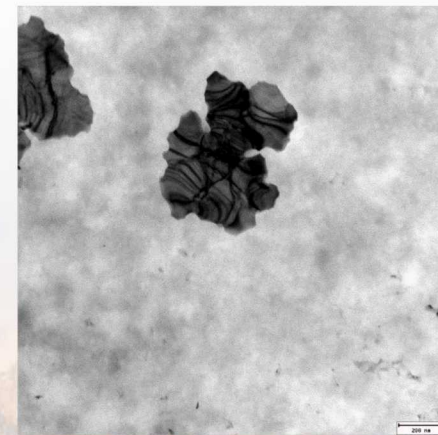
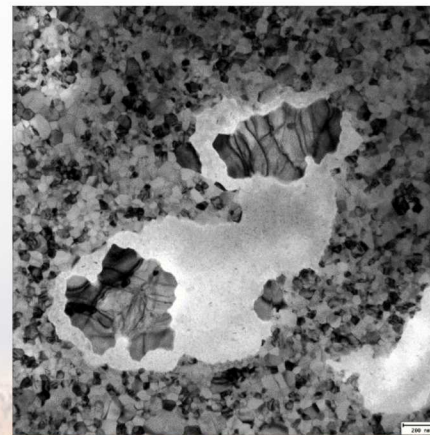
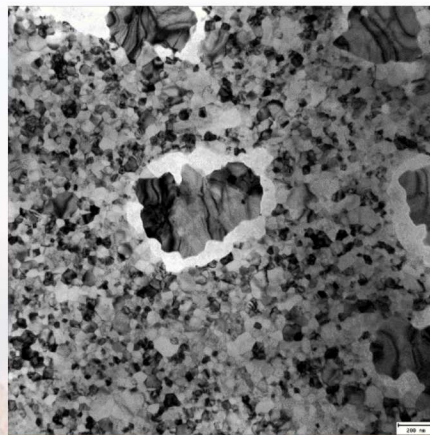
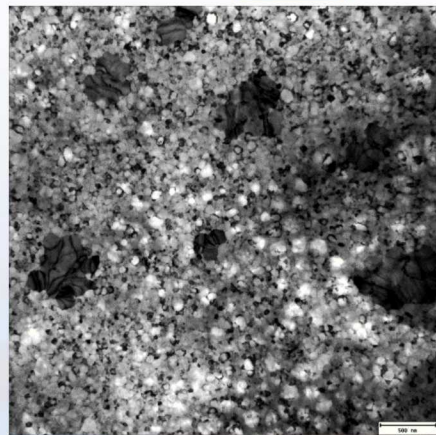
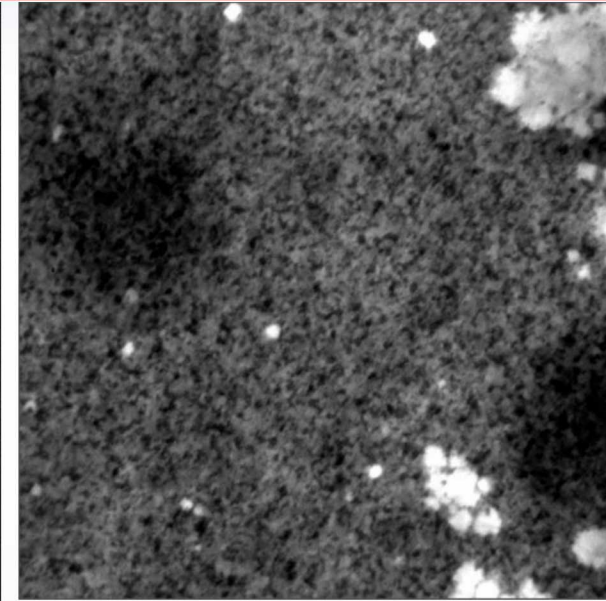
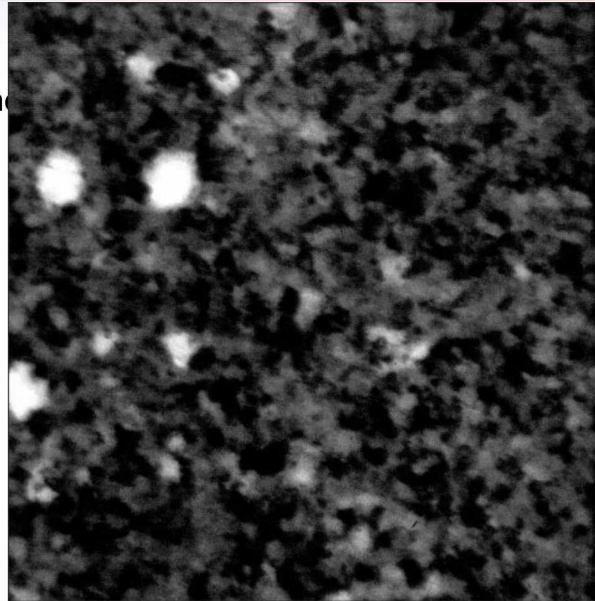
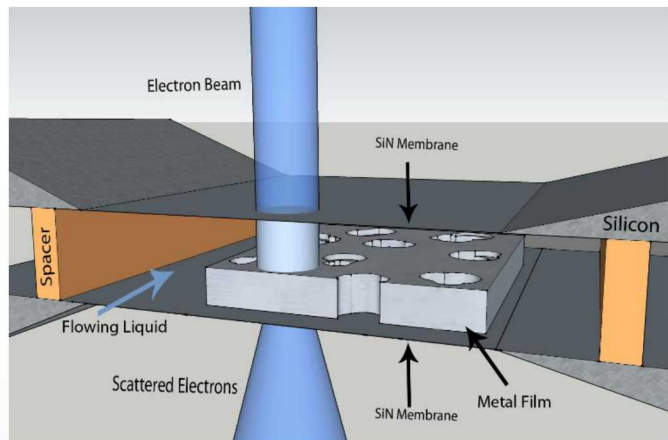


# Can We Gain Insight into the Corrosion Process through *In situ* TEM?

Contributors: D. Gross, J. Kacher, I.M. Robertson & Protochips, Inc.

## Microfluidic Stage

- Mixing of two or more channels
- Continuous observation of the reaction channels



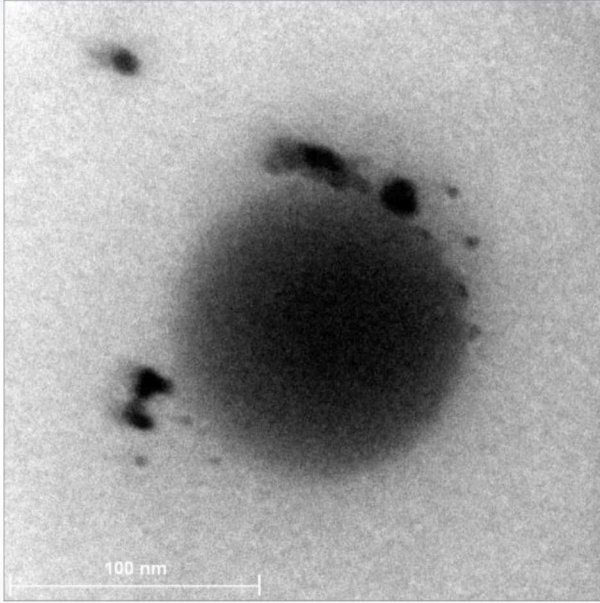
Pitting mechanisms during dilute flow of acetic acid over 99.95% nc-PLD Fe involves many grains. Large grains resulting from annealing appear more corrosion tolerant

# Other Fun Uses of Microfluidic Cell

## Protocell Drug Delivery

S. Hoppe,  
E. Carnes,  
J. Brinker

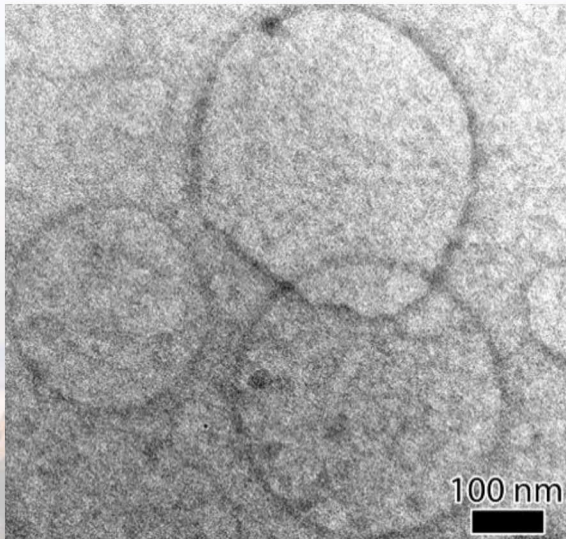
Liposome  
encapsulated  
Silica destroyed  
by the electron  
beam



## Liposomes in Water

S. Hoppe,  
D. Sasaki

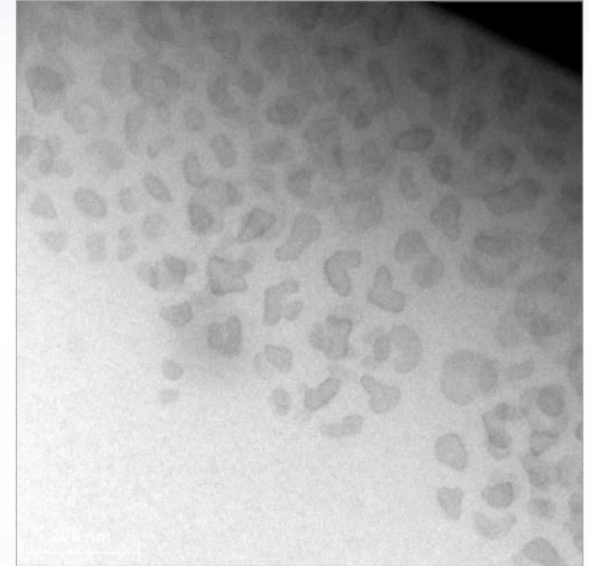
Liposomes  
imaged in  
flowing aqueous  
channel



## BSA Crystallization

S. Hoppe

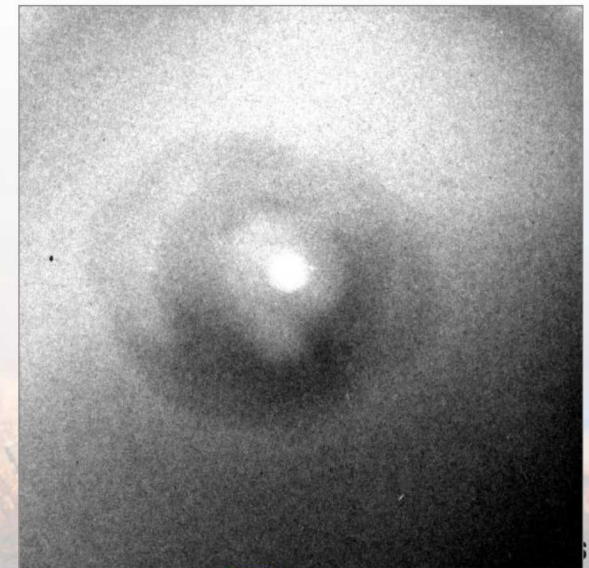
Crystallization of excess  
Bovine Serum Albumen  
during flow



## La Structure Formation

S. Hoppe,  
T. Nenoff

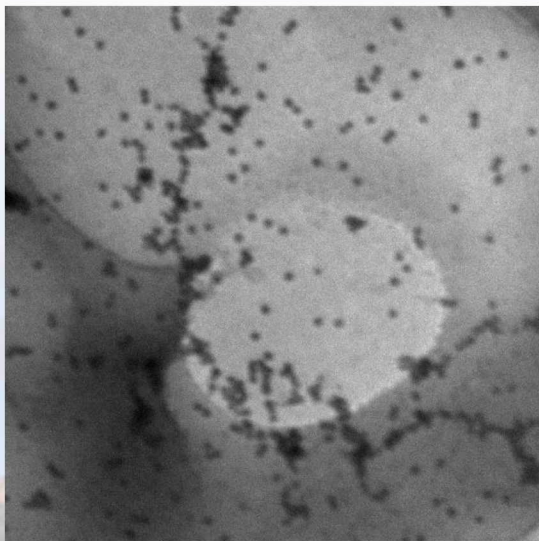
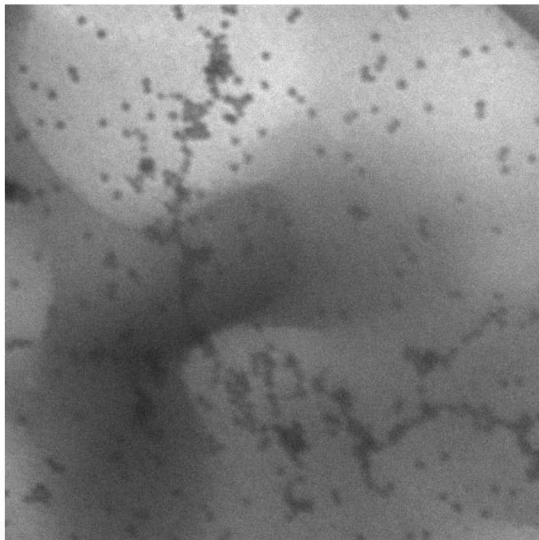
La  
Nanostructure  
form from  $\text{LaCl}_3$   
 $\text{H}_2\text{O}$  in wet cell  
due to beam  
effects



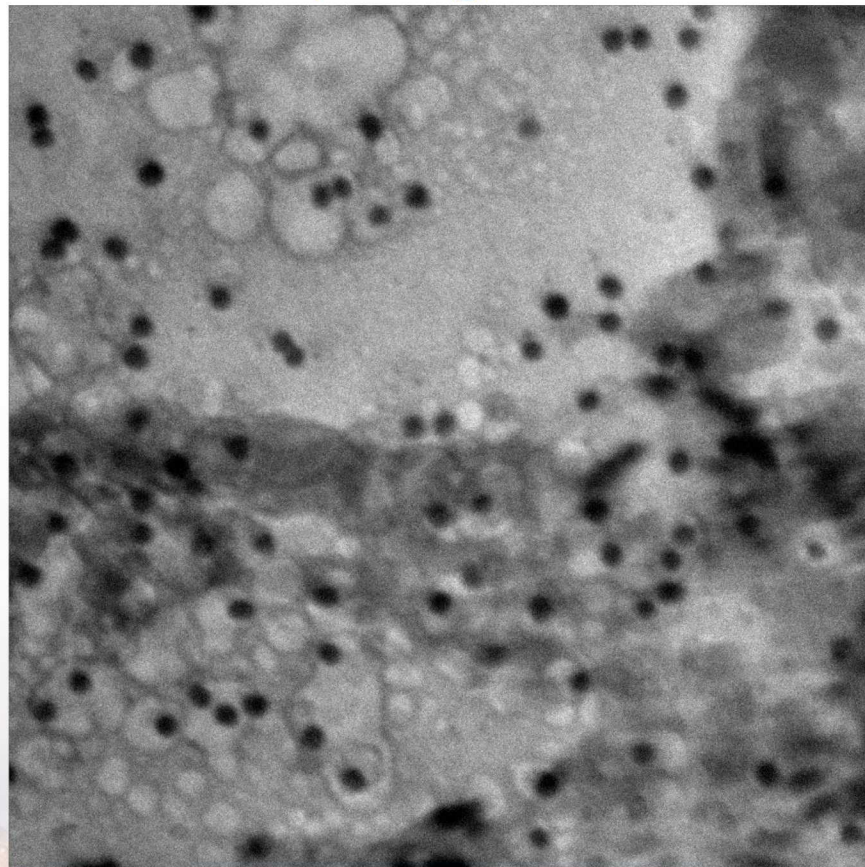
# *In Situ* TEM observation of three-phase mixture

Collaborator: Sarah Pratt

## Au nanoparticles, aqueous solution, and gas bubbles



A pair of images  
before and after  
radiolysis from  
focusing the  
electron beam



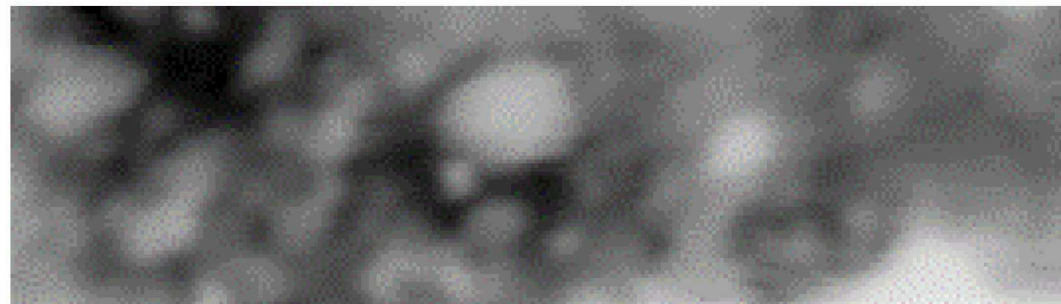
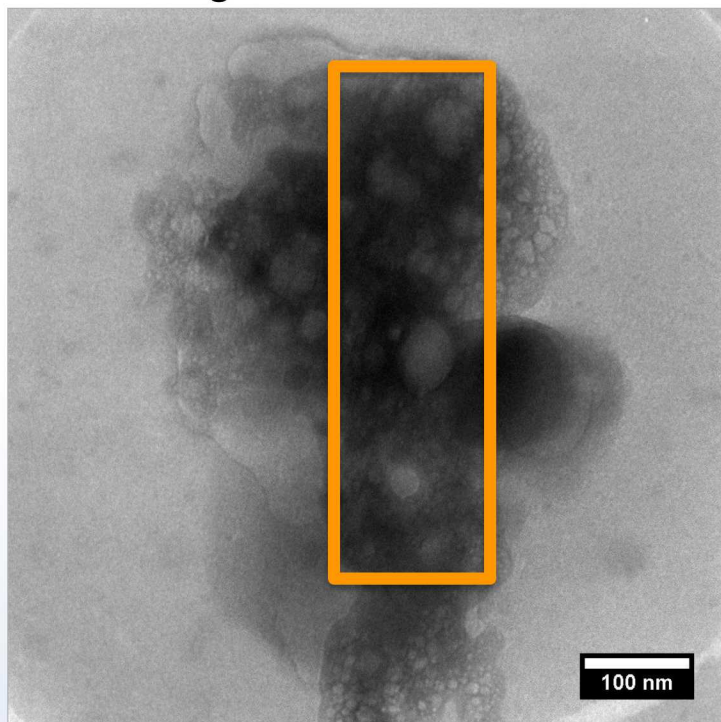
# PED in Liquid Cell Environment

Contributors: C. Taylor, S. Pratt, & T. Nenoff

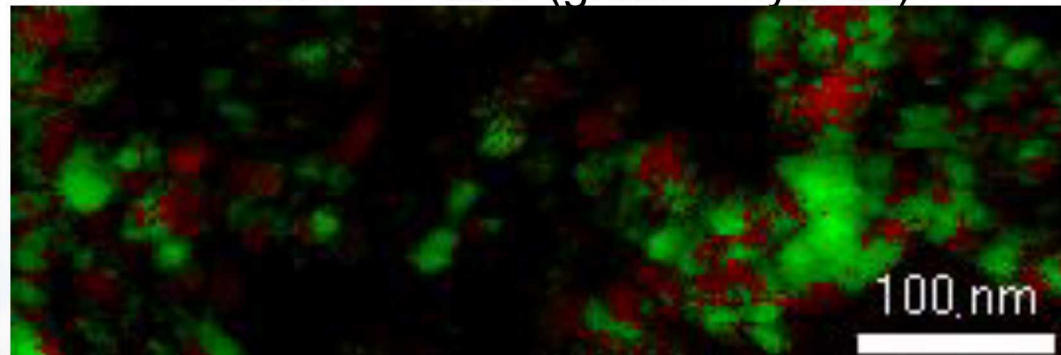


Virtual BF

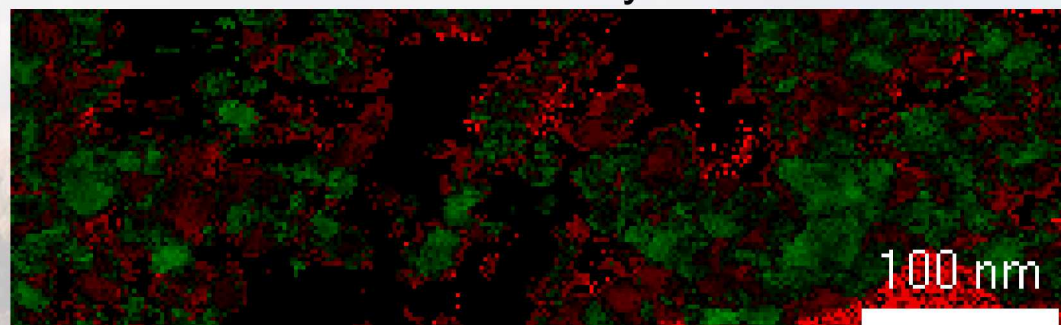
TEM Image



Index + Phase (green = hydride)



Phase Reliability + Phase

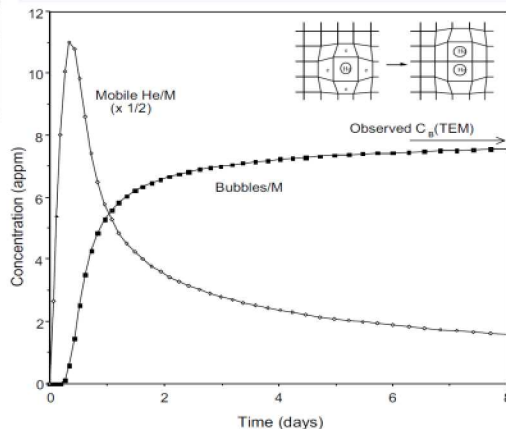
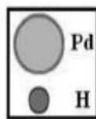
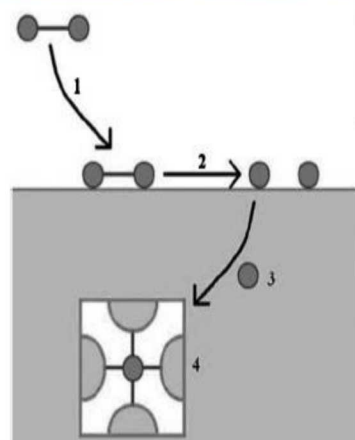


Virtual BF



# In situ TEM Hydrogen Exposure

Contributors: B.G. Clark, P.J. Cappillino, B.W. Jacobs, M.A. Hekmaty, D.B. Robinson, L.R. Parent, I. Arslan, & Protochips, Inc.



R. Delmelle, J., Phys. Chem. Chem. Phys. (2011) p.11412

Cowgill, D., Fusion Sci. & Tech., 28 (2005) p. 539

Trinkaus, H. et al., JNM (2003) p. 229

Thiebaut, S. et al. JNM (2000) p. 217

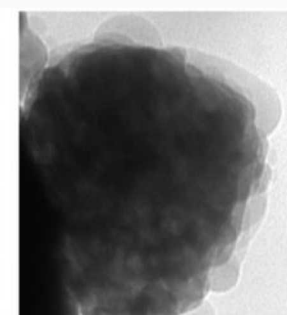
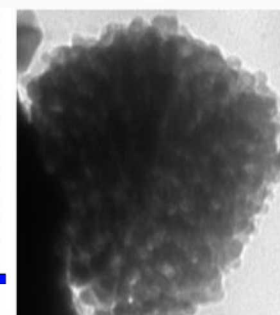
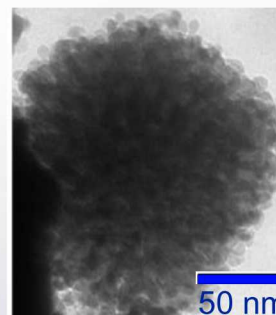
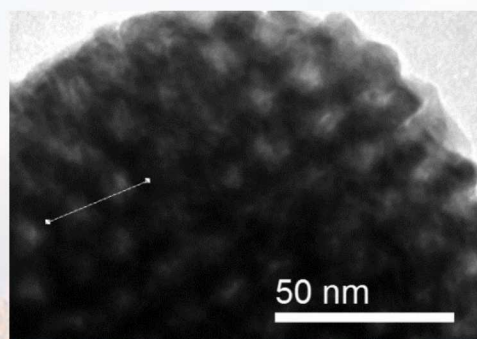
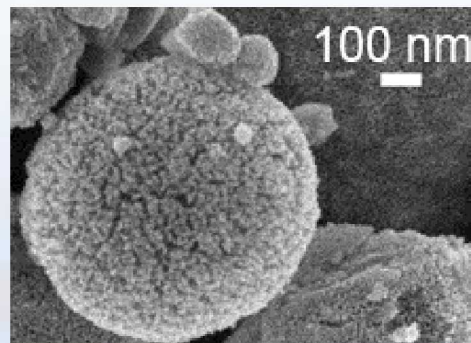
## Vapor-Phase Heating TEM Stage

- Compatible with a range of gases
- In situ* resistive heating
- Continuous observation of the reaction channel
- Chamber dimensions are controllable
- Compatible with MS and other analytical tools



- 1 atm  $H_2$  after several pulses to specified temp.

Harmful effects may be mitigated in nanoporous Pd



125°

200°

300°

New *in situ* atmospheric heating experiments provide great insight into nanoporous Pd stability



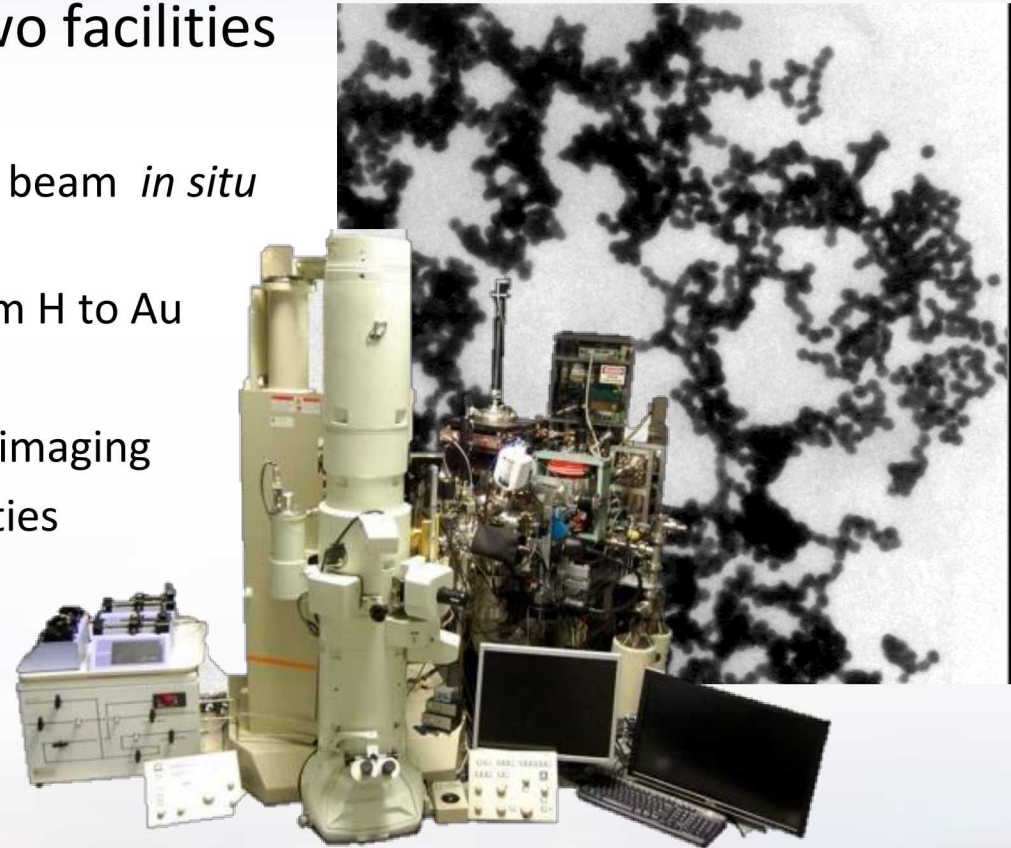


# Unconventional *In situ* Microscopy Creates a wealth of Possibilities



- Sandia's I<sup>3</sup>TEM is one of only two facilities in the US
  - Only facility in the world with a triple beam *in situ* ion irradiation capabilities
  - *In situ* high energy ion irradiation from H to Au
  - *In situ* gas implantation
  - Dynamic TEM and time compression imaging
  - 11+ TEM stages with various capabilities

Currently applying the I<sup>3</sup>TEM capabilities to various material systems in combined environmental conditions



## Collaborators:

D.L. Buller, D.C. Bufford, S.H. Pratt, T.J. Boyle, B.A. Hernandez-Sanchez, S.J. Blair, B. Muntiferung, C. Chisholm, P. Hosemann, A. Minor, J. A. Hinks, F. Hibberd, A. Ilinov, D. C. Bufford, F. Djurabekova, G. Greaves, A. Kuronen, S. E. Donnelly, K. Nordlund, F. Abdeljawad, S.M. Foiles, J. Qu, C. Taylor, J. Sugar, P. Price, C.M. Barr, D. Adams, M. Abere, L. Treadwell, A. Cook, A. Monterrosa, IDES Inc, J. Sharon, B. L. Boyce, C. Chisholm, H. Bei, E.P. George, W. Mook, Hysitron Inc., G.S. Jawaharram, S. Dillon, R.S. Averbach, N. Heckman, J. Carroll, S. Briggs, E. Carnes, J. Brinker, D. Sasaki, T. Nenoff, B.G. Clark, P.J. Cappillino, B.W. Jacobs, M.A. Hekmaty, D.B. Robinson, L.R. Parent, I. Arslan, & Protochips

This work was partially funded by the Division of Materials Science and Engineering, Office of Basic Energy Sciences, U.S. Department of Energy. Materials Science and Engineering, Office of Basic Energy Sciences, U.S. Department of Energy. This work was performed, in part, at the Center for Integrated Nanotechnologies, an Office of Science User Facility operated for the U.S. Department of Energy (DOE) Office of Science. Sandia National Laboratories is a multi-mission laboratory managed and operated by National Technology and Engineering Solutions of Sandia, LLC., a wholly owned subsidiary of Honeywell International, Inc., for the U.S. DOE's National Nuclear Security Administration under contract DE-NA-0003525. The views expressed in the article do not necessarily represent the views of the U.S. DOE or the United States Government.



U.S. DEPARTMENT OF  
**ENERGY**

Office of  
Science



Sandia National Laboratories



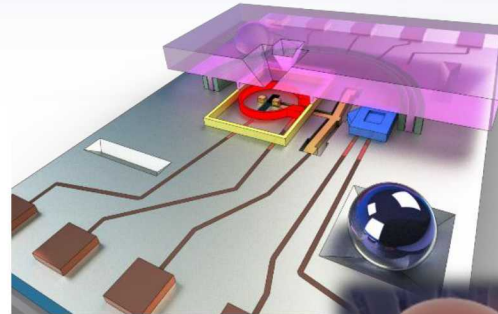
# Sandia's USER Capabilities



D. Hanson, W. Martin, M. Wasiolek

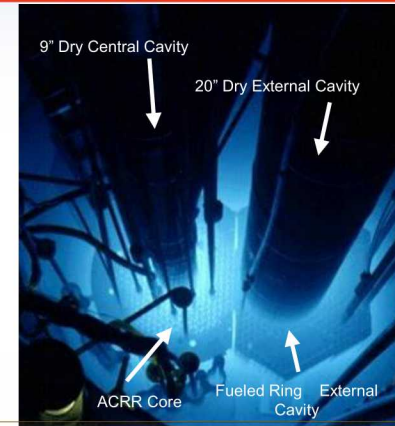
[www.cint.lanl.gov](http://www.cint.lanl.gov)

- Spring and Fall proposals for 18 months
- Rapid Access proposal anytime for 3 months



[www.nsunf.inl.gov](http://www.nsunf.inl.gov)

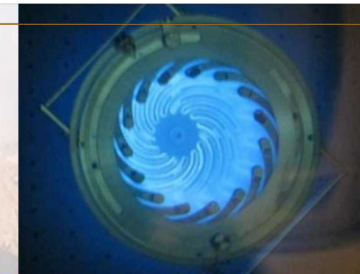
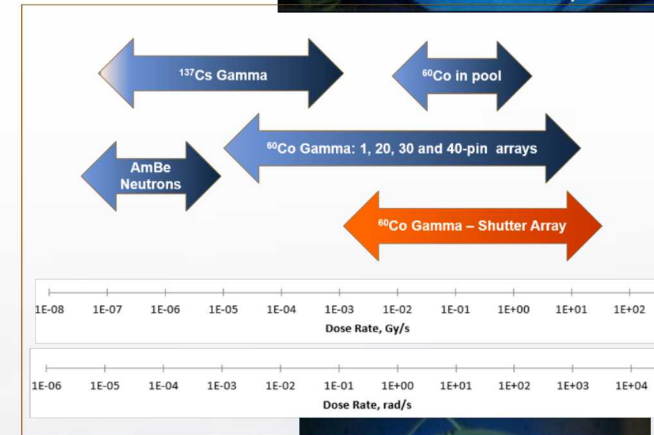
- Three proposal a year for 9 months



## Core Facility - SNL



## Gateway Facility - LANL



This work was partially funded by the Division of Materials Science and Engineering, Office of Basic Energy Sciences, U.S. Department of Energy. Materials Science and Engineering, Office of Basic Energy Sciences, U.S. Department of Energy. This work was performed, in part, at the Center for Integrated Nanotechnologies, an Office of Science User Facility operated for the U.S. Department of Energy (DOE) Office of Science. Sandia National Laboratories is a multi-mission laboratory managed and operated by National Technology and Engineering Solutions of Sandia, LLC., a wholly owned subsidiary of Honeywell International, Inc., for the U.S. DOE's National Nuclear Security Administration under contract DE-NA-0003525. The views expressed in the article do not necessarily represent the views of the U.S. DOE or the United States Government.

