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Peridynamic model for fatigue cracks

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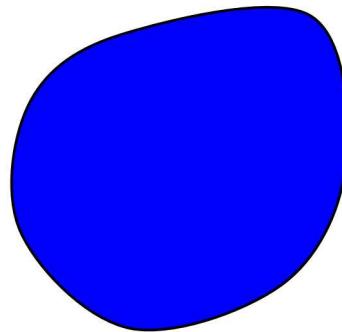
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Outline

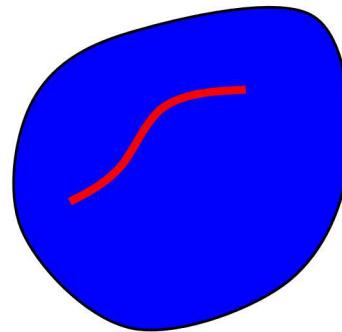
- Peridynamics: What it is
- Damage and fracture
- Fatigue model
- Calibration of parameters

Purpose of peridynamics*

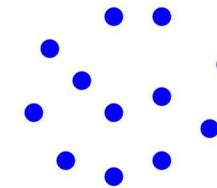
- To unify the mechanics of continuous and discontinuous media within a single, consistent set of equations.



Continuous body



Continuous body
with a defect



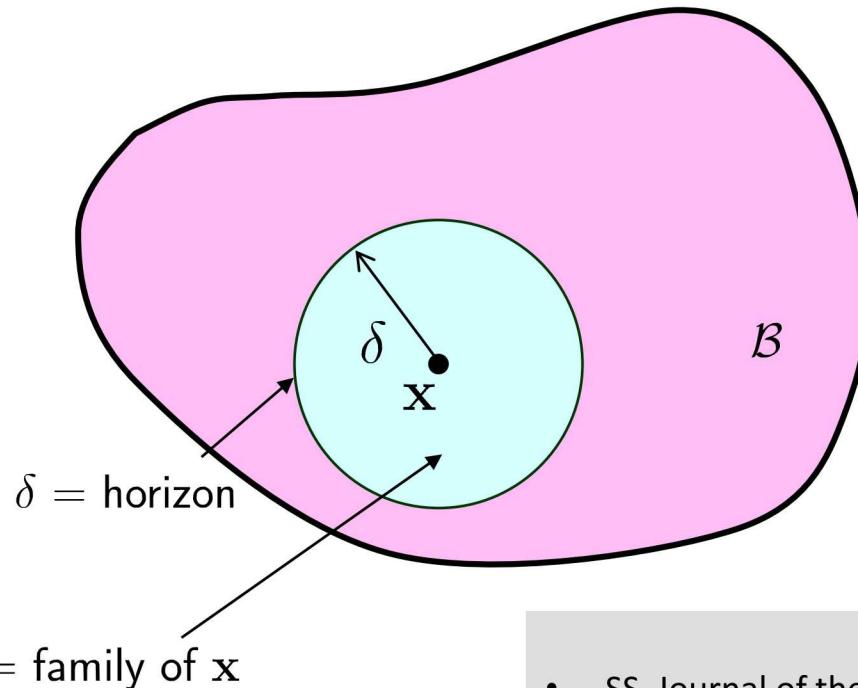
Discrete particles

- Why do this?
 - Avoid coupling dissimilar mathematical systems (A to C).
 - Model complex fracture patterns.
 - Communicate across length scales.

* Peri (near) + dyn (force)

Peridynamics basics: Horizon and family

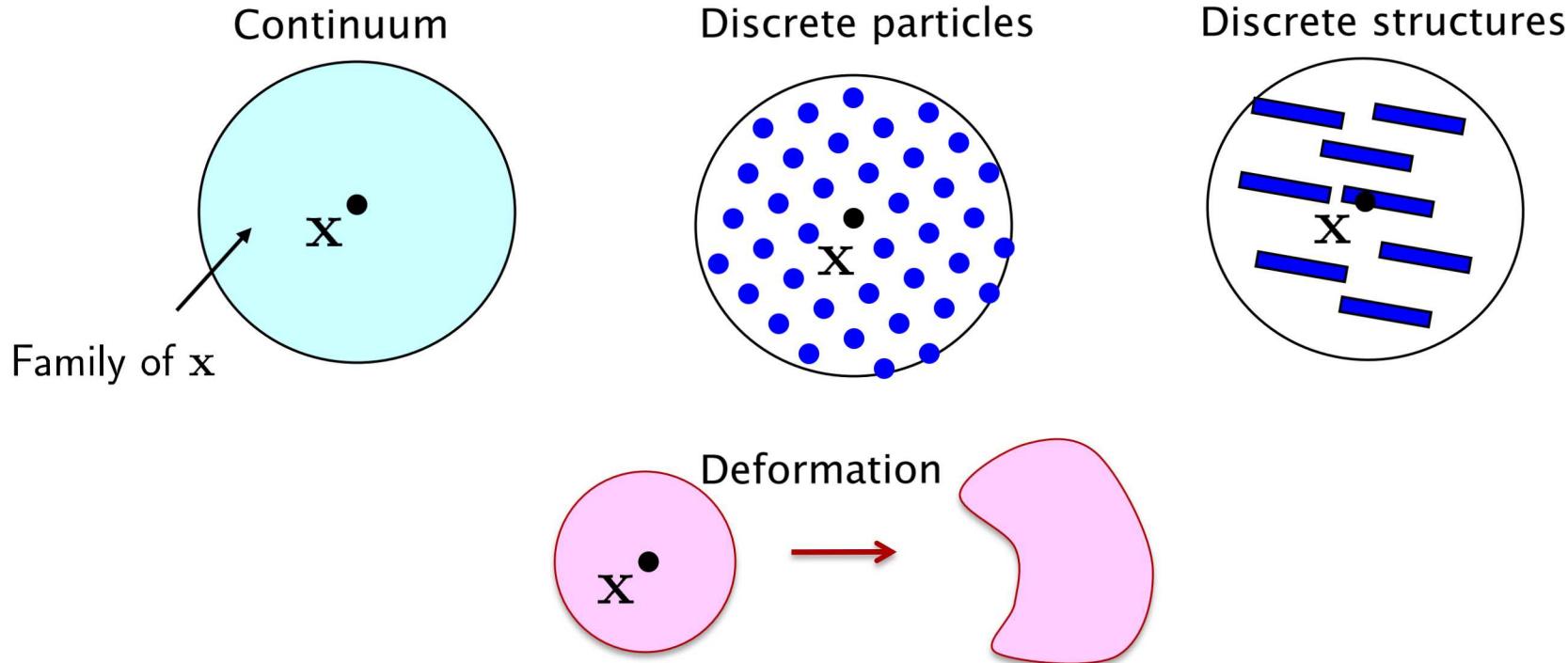
- Any point x interacts directly with other points within a distance δ called the “horizon.”
- The material within a distance δ of x is called the “family” of x , \mathcal{H}_x .



General references

- Silling, *Journal of the Mechanics and Physics of Solids* (2000)
- Silling and R. Lehoucq, *Advances in Applied Mechanics* (2010)
- Madenci & Oterkus, *Peridynamic Theory & Its Applications* (2014)

Peridynamic nonlocality: Strain energy at a point



- Key assumption: the strain energy density at $W(x)$ is determined by the deformation of its family.

Potential energy minimization yields the peridynamic equilibrium equation

- Potential energy:

$$\Phi = \int_{\mathcal{B}} (W - \mathbf{b} \cdot \mathbf{y}) \, dV_{\mathbf{x}}$$

where W is the strain energy density, \mathbf{y} is the deformation map, \mathbf{b} is the applied external force density, and \mathcal{B} is the body.

- Euler-Lagrange equation is the equilibrium equation:

$$\int_{\mathcal{H}_{\mathbf{x}}} \mathbf{f}(\mathbf{q}, \mathbf{x}) \, dV_{\mathbf{q}} + \mathbf{b}(\mathbf{x}) = 0$$

for all \mathbf{x} . \mathbf{f} is the *pairwise bond force density*.

Peridynamics basics: States

- A *peridynamic state* is a mapping on bonds in a family.
- We write:

$$\mathbf{u} = \underline{\mathbf{A}} \langle \xi \rangle$$

where ξ is a bond, $\underline{\mathbf{A}}$ is a state, and \mathbf{u} is some vector.

- States play a role in peridynamics similar to that of second order tensors in the local theory.

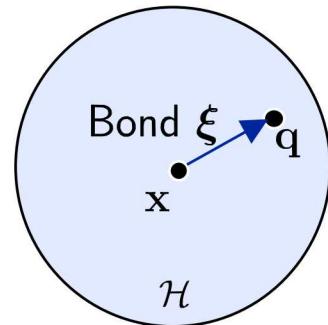
Peridynamics basics: Kinematics

- The *deformation state* is the function that maps each bond ξ into its deformed image:

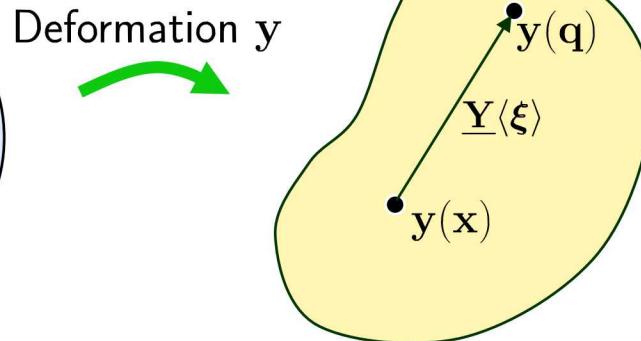
$$\underline{Y}(\xi) = \mathbf{y}(\mathbf{q}) - \mathbf{y}(\mathbf{x})$$

where \mathbf{y} is the deformation and

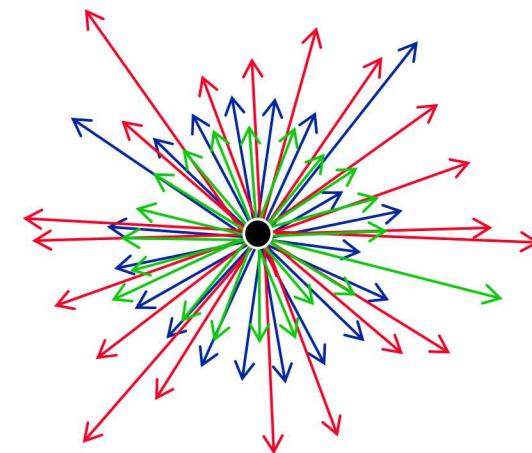
$$\xi = \mathbf{q} - \mathbf{x}.$$



Undeformed family of \mathbf{x}



Deformed family of \mathbf{x}



Deformed images of bonds:
State description allows complexity

Peridynamics basics: Force state

- $\mathbf{f}(\mathbf{x}, \mathbf{q})$ has contributions from the material models at both \mathbf{x} and \mathbf{q} .

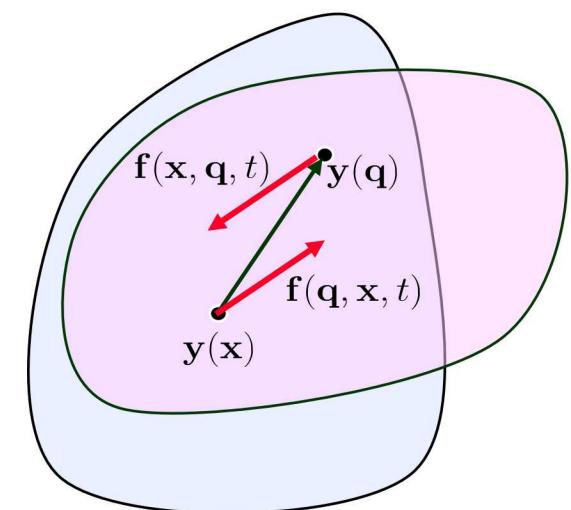
$$\mathbf{f}(\mathbf{x}, \mathbf{q}) = \mathbf{t}(\mathbf{x}, \mathbf{q}) - \mathbf{t}(\mathbf{q}, \mathbf{x})$$

$$\mathbf{t}(\mathbf{x}, \mathbf{q}) = \underline{\mathbf{T}}[\mathbf{x}] \langle \mathbf{q} - \mathbf{x} \rangle, \quad \mathbf{t}(\mathbf{x}, \mathbf{q}) = \underline{\mathbf{T}}[\mathbf{q}] \langle \mathbf{x} - \mathbf{q} \rangle$$

- $\underline{\mathbf{T}}[\mathbf{x}]$ is the *force state*: maps bonds onto bond force densities. It is found from the constitutive model:

$$\underline{\mathbf{T}} = \hat{\underline{\mathbf{T}}}(\underline{\mathbf{Y}})$$

where $\hat{\underline{\mathbf{T}}}$ maps the deformation state to the force state.



Peridynamics basics: Bonds and bond force density

- The vector from \mathbf{x} to any point \mathbf{q} in its family in the reference configuration is called a *bond*.

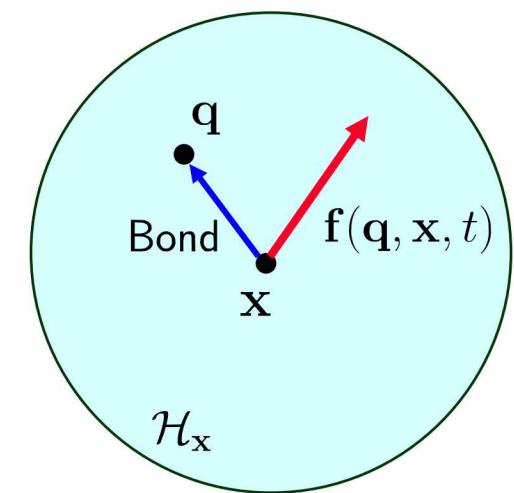
$$\boldsymbol{\xi} = \mathbf{q} - \mathbf{x}$$

- Each bond has a *pairwise force density* vector that is applied at both points:

$$\mathbf{f}(\mathbf{q}, \mathbf{x}, t).$$

- Equation of motion is an integro-differential equation, not a PDE:

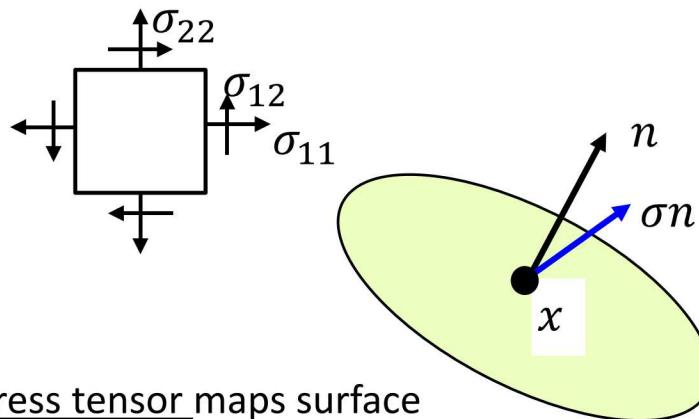
$$\rho(\mathbf{x})\ddot{\mathbf{y}}(\mathbf{x}, t) = \int_{\mathcal{H}} \mathbf{f}(\mathbf{q}, \mathbf{x}, t) \, dV_{\mathbf{q}} + \mathbf{b}(\mathbf{x}, t).$$



Peridynamics basics: The nature of internal forces

Standard theory

Stress tensor field
(assumes continuity of forces)



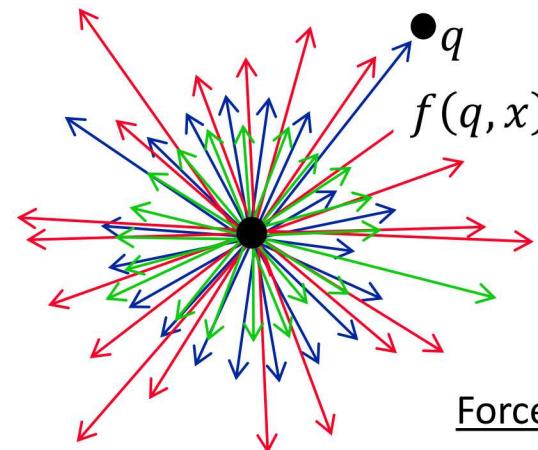
Stress tensor maps surface
normal vectors onto
surface forces

$$\rho \ddot{u}(x, t) = \nabla \cdot \sigma(x, t) + b(x, t)$$

Differentiation of surface forces

Peridynamics

Bond forces between neighboring points
(allowing discontinuity)



Force state maps bonds
onto bond forces

$$\rho \ddot{u}(x, t) = \int_{H_x} f(q, x) dV_q + b(x, t)$$

Summation over bond forces

Peridynamics basics: Elastic materials

- A peridynamic elastic material has strain energy density given by

$$W(\underline{\mathbf{Y}}).$$

- The force state is given by

$$\hat{\underline{\mathbf{T}}}(\underline{\mathbf{Y}}) = W_{\underline{\mathbf{Y}}}(\underline{\mathbf{Y}})$$

where $W_{\underline{\mathbf{Y}}}$ is the Frechet derivative of the strain energy density.

Peridynamic vs. local equations

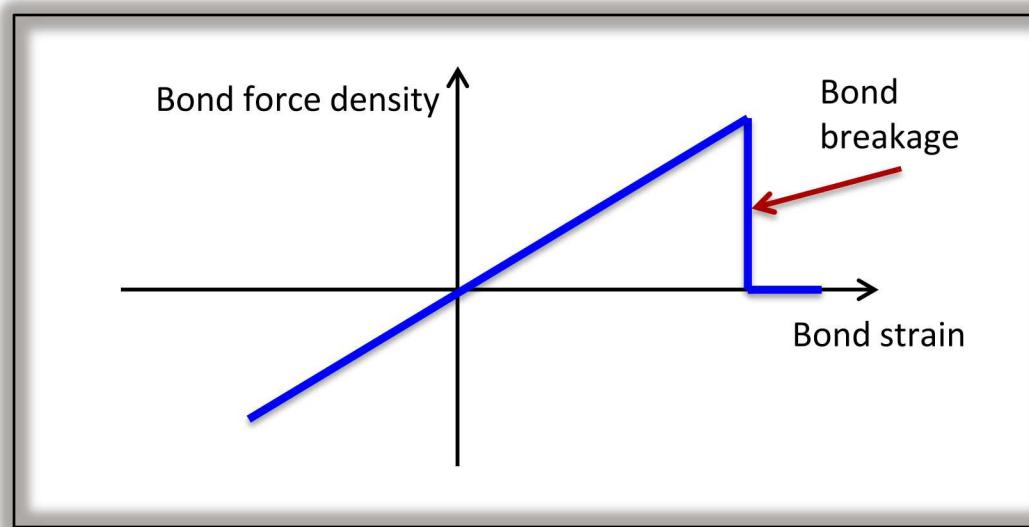
- The structures of the theories are similar, but peridynamics uses nonlocal operators.

Relation	<i>Peridynamic theory</i>	<i>Standard theory</i>
Kinematics	$\underline{\mathbf{Y}}\langle \mathbf{q} - \mathbf{x} \rangle = \mathbf{y}(\mathbf{q}) - \mathbf{y}(\mathbf{x})$	$\mathbf{F}(\mathbf{x}) = \frac{\partial \mathbf{y}}{\partial \mathbf{x}}(\mathbf{x})$
Linear momentum balance	$\rho \ddot{\mathbf{y}}(\mathbf{x}) = \int_{\mathcal{H}} \left(\mathbf{t}(\mathbf{q}, \mathbf{x}) - \mathbf{t}(\mathbf{x}, \mathbf{q}) \right) dV_{\mathbf{q}} + \mathbf{b}(\mathbf{x})$	$\rho \ddot{\mathbf{y}}(\mathbf{x}) = \nabla \cdot \boldsymbol{\sigma}(\mathbf{x}) + \mathbf{b}(\mathbf{x})$
Constitutive model	$\mathbf{t}(\mathbf{q}, \mathbf{x}) = \underline{\mathbf{T}}\langle \mathbf{q} - \mathbf{x} \rangle, \quad \underline{\mathbf{T}} = \hat{\underline{\mathbf{T}}}(\underline{\mathbf{Y}})$	$\boldsymbol{\sigma} = \hat{\boldsymbol{\sigma}}(\mathbf{F})$
Angular momentum balance	$\int_{\mathcal{H}} \underline{\mathbf{Y}}\langle \mathbf{q} - \mathbf{x} \rangle \times \underline{\mathbf{T}}\langle \mathbf{q} - \mathbf{x} \rangle dV_{\mathbf{q}} = \mathbf{0}$	$\boldsymbol{\sigma} = \boldsymbol{\sigma}^T$
Elasticity	$\underline{\mathbf{T}} = W_{\underline{\mathbf{Y}}} \text{ (Fréchet derivative)}$	$\boldsymbol{\sigma} = W_{\mathbf{F}} \text{ (tensor gradient)}$
First law	$\dot{\varepsilon} = \underline{\mathbf{T}} \bullet \dot{\underline{\mathbf{Y}}} + q + r$	$\dot{\varepsilon} = \boldsymbol{\sigma} \cdot \dot{\mathbf{F}} + q + r$

$$\underline{\mathbf{T}} \bullet \dot{\underline{\mathbf{Y}}} := \int_{\mathcal{H}} \underline{\mathbf{T}}\langle \xi \rangle \cdot \dot{\underline{\mathbf{Y}}}\langle \xi \rangle dV_{\xi}$$

Bond based material models

- If each bond response is independent of the others, the resulting material model is called bond-based.
- The material model is then simply a graph of bond force density vs. bond strain.
- Damage can be modeled through bond breakage.
- Bond response is calibrated to:
 - Bulk elastic properties.
 - Critical energy release rate.



EMU numerical method

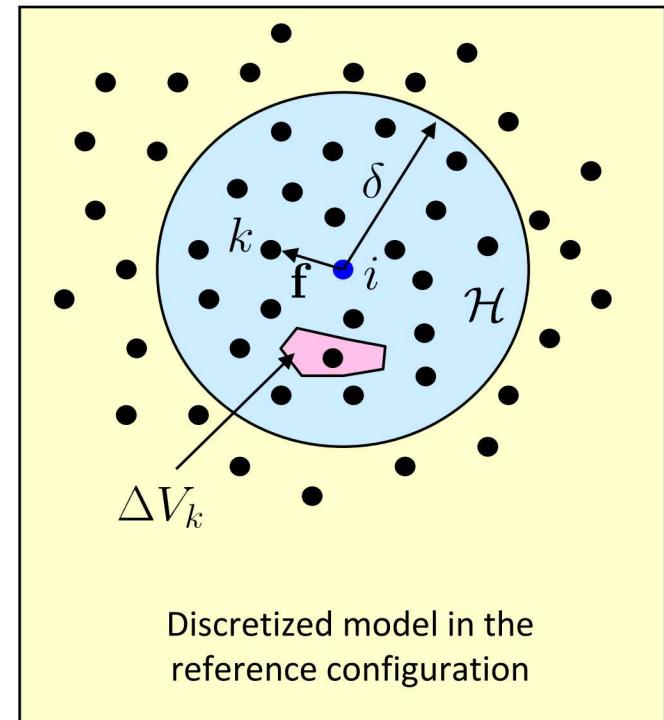
- Integral is replaced by a finite sum: resulting method is meshless and Lagrangian.

$$\rho \ddot{\mathbf{y}}(\mathbf{x}, t) = \int_{\mathcal{H}} \mathbf{f}(\mathbf{q}, \mathbf{x}, t) \, dV_{\mathbf{q}} + \mathbf{b}(\mathbf{x}, t)$$

↓

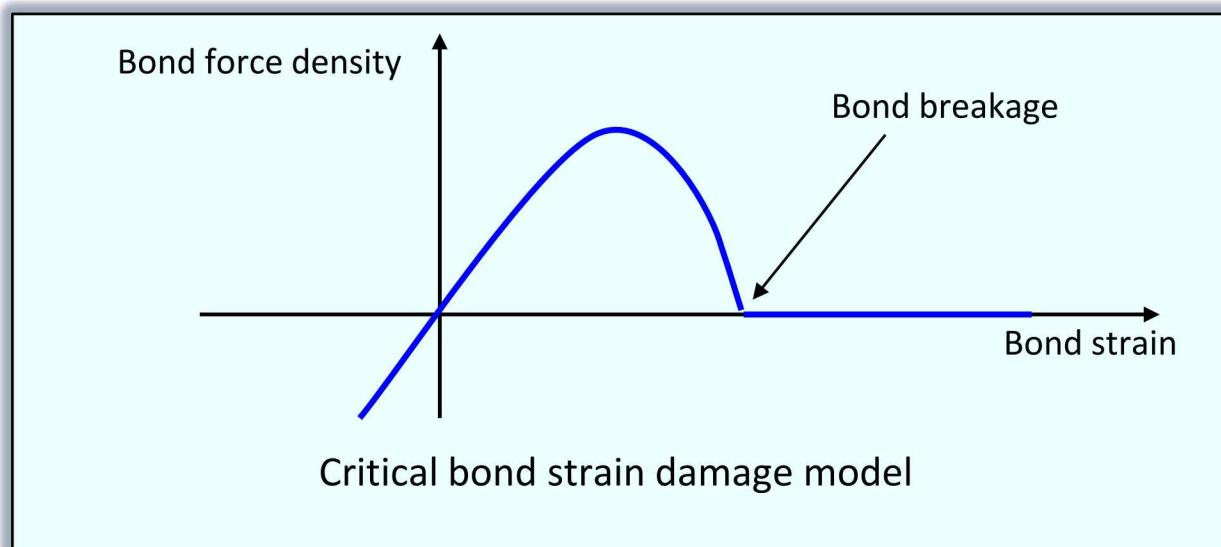
$$\rho \ddot{\mathbf{y}}_i^n = \sum_{k \in \mathcal{H}} \mathbf{f}(\mathbf{x}_k, \mathbf{x}_i, t) \, \Delta V_k + \mathbf{b}_i^n$$

- Looks a lot like MD.
- Unrelated to Smoothed Particle Hydrodynamics
 - SPH solves the local equations by fitting spatial derivatives to the current node values.

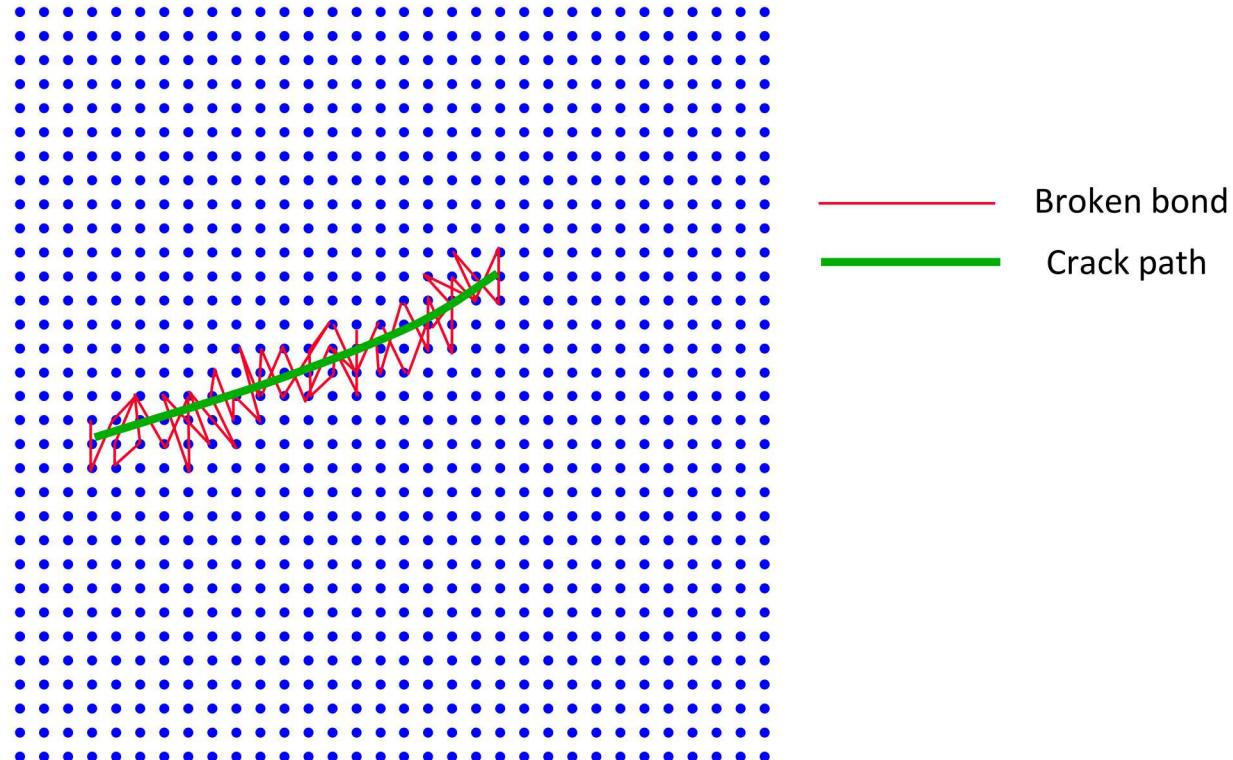


Damage due to bond breakage

- Recall: each bond carries a force.
- Damage is implemented at the bond level.
 - Bonds break irreversibly according to some criterion.
 - Broken bonds carry no force.
- Examples of criteria:
 - Critical bond strain (brittle).
 - Hashin failure criterion (composites).
 - Gurson (ductile metals).



Autonomous crack growth

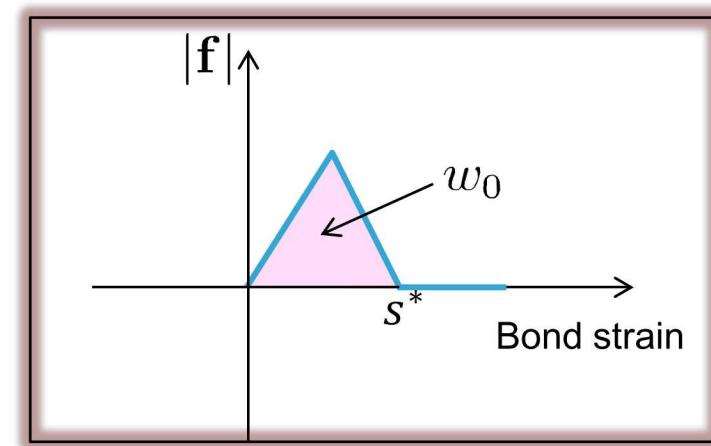
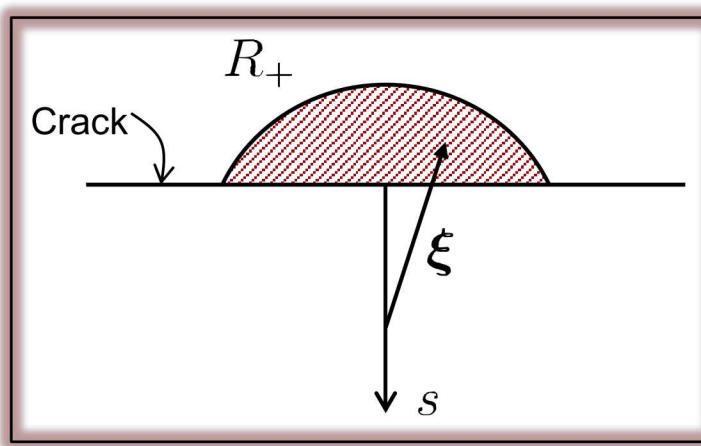


- When a bond breaks, its load is shifted to its neighbors, leading to progressive failure.

Critical bond strain: Relation to critical energy release rate

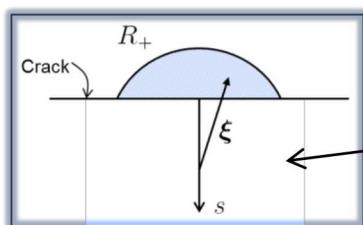
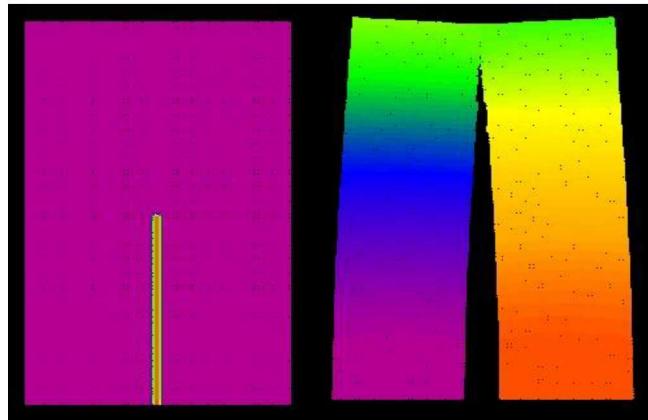
If the work required to break the bond ξ is $w_0(\xi)$, then the energy release rate is found by summing this work per unit crack area (J. Foster):

$$G = \int_0^\delta \int_{R_+} w_0(\xi) dV_\xi ds$$



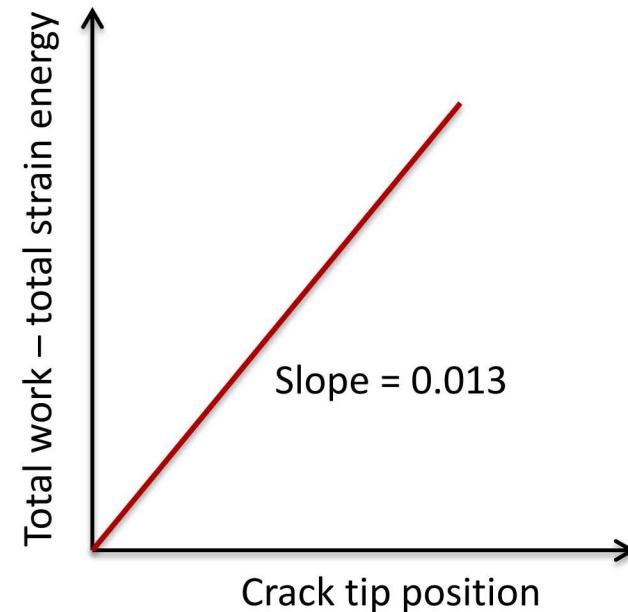
- Can then get the critical strain for bond breakage s^* in terms of G .
- Could also use the peridynamic J-integral as a bond breakage criterion.

Constant bond failure strain reproduces the Griffith crack growth criterion



From bond properties, energy release rate should be

$$G = 0.013$$



- This confirms that the energy consumed per unit crack growth area equals the expected value from bond breakage properties.

Some results about peridynamics

- For any choice of horizon, we can fit material model parameters to match the bulk properties and energy release rate.
 - Using nonlocality, can obtain material model parameters from wave dispersion curves (Weckner).
- Coupled coarse scale and fine scale evolution equations can be derived for composites (Lipton and Alali).
- A set of discrete particles interacting through any multibody potential can be represented exactly as a peridynamic body.
- Well posedness has been established under certain conditions (Mangesh, Du, Gunzburger, Lehoucq).

Cyclic strain in a bond

- For a given bond ξ , the *bond elongation* is the change in bond length:

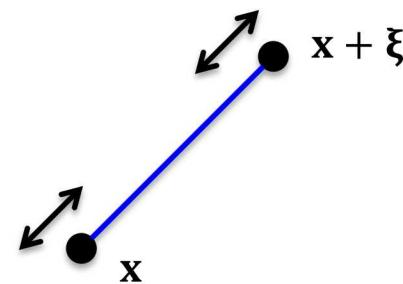
$$e = |\mathbf{Y}\langle \xi \rangle| - |\xi| = |\mathbf{y}(x + \xi) - \mathbf{y}(x)|.$$

- The *bond strain* is the change in length over initial length:

$$s = \frac{e}{|\xi|}.$$

- Let s^+ and s^- be the two extremes under cyclic loading of ξ .
- The *cyclic bond strain* is defined by

$$\varepsilon = |s^+ - s^-|.$$

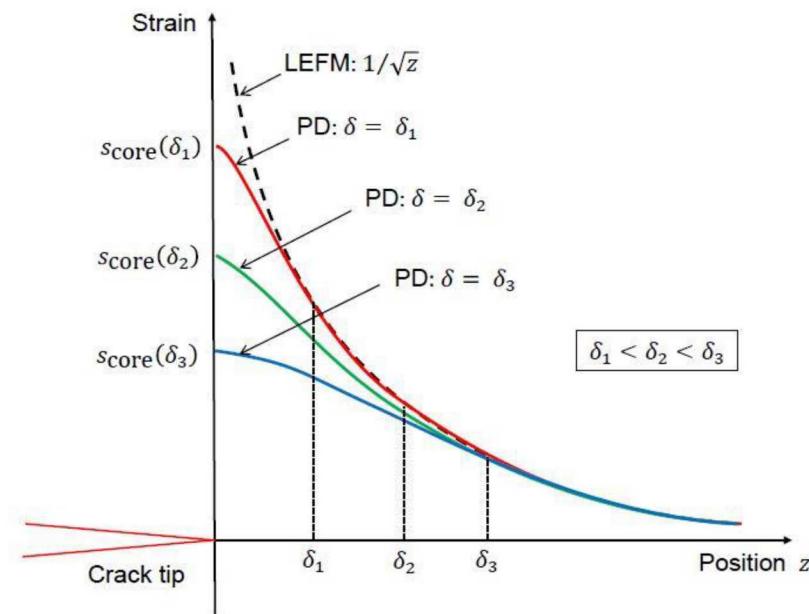
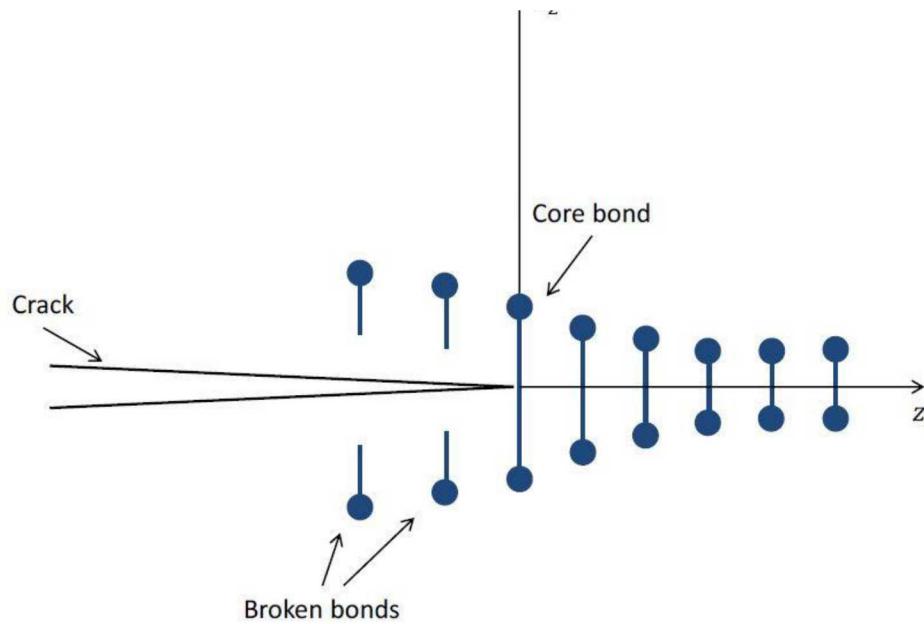


Structure of a crack tip field

- Let $\varepsilon_{\text{core}}(\delta)$ be the largest cyclic strain in any bond.
- Can show by a dimensional argument $\exists \hat{\varepsilon}_{\text{core}}$ such that

$$\varepsilon_{\text{core}}(\delta) = \hat{\varepsilon}_{\text{core}} \frac{\Delta K}{E\sqrt{\delta}}$$

where ΔK = cyclic stress intensity factor and E = modulus.



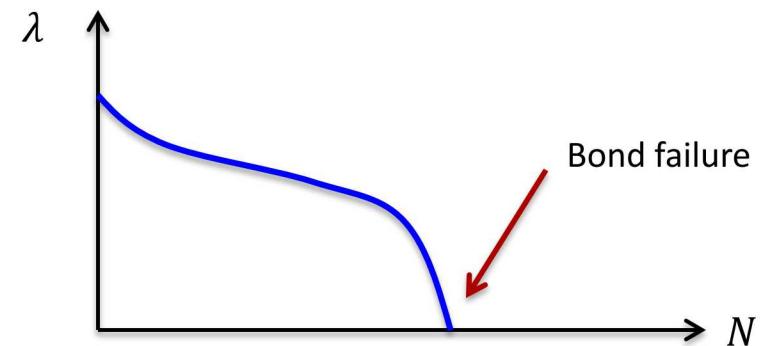
Remaining life of a bond

- Each bond in the body has a *remaining life* $\lambda(N)$ where N is the cycle number.
- The remaining life is monotonically decreasing over time.

$$\lambda(0) = 1, \quad \dot{\lambda} \leq 0.$$

- The bond fails at the first cycle N when

$$\lambda(N) \leq 0.$$



Fatigue model

- The fatigue model specifies how the remaining life of each bond depends on the loading.

$$\frac{d\lambda}{dN}(N) = -A\varepsilon^m$$

where A and m are constants and ε is the cyclic bond strain.

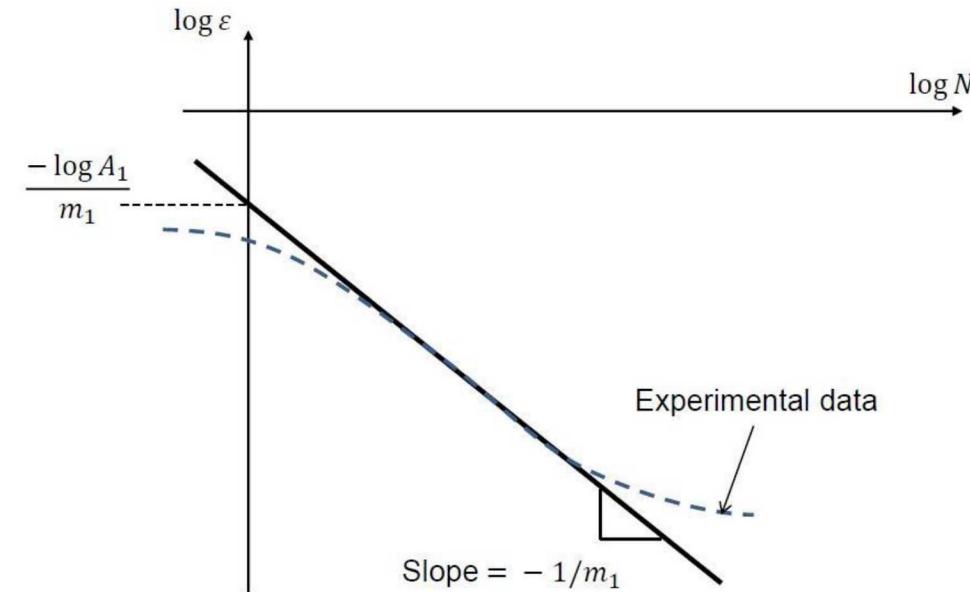
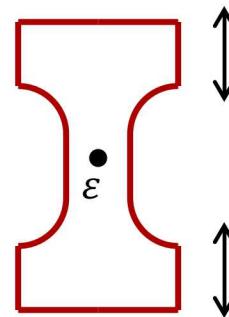
- The constants are calibrated separately for phases I and II (nucleation and growth).

Phase I calibration from $S-N$ data

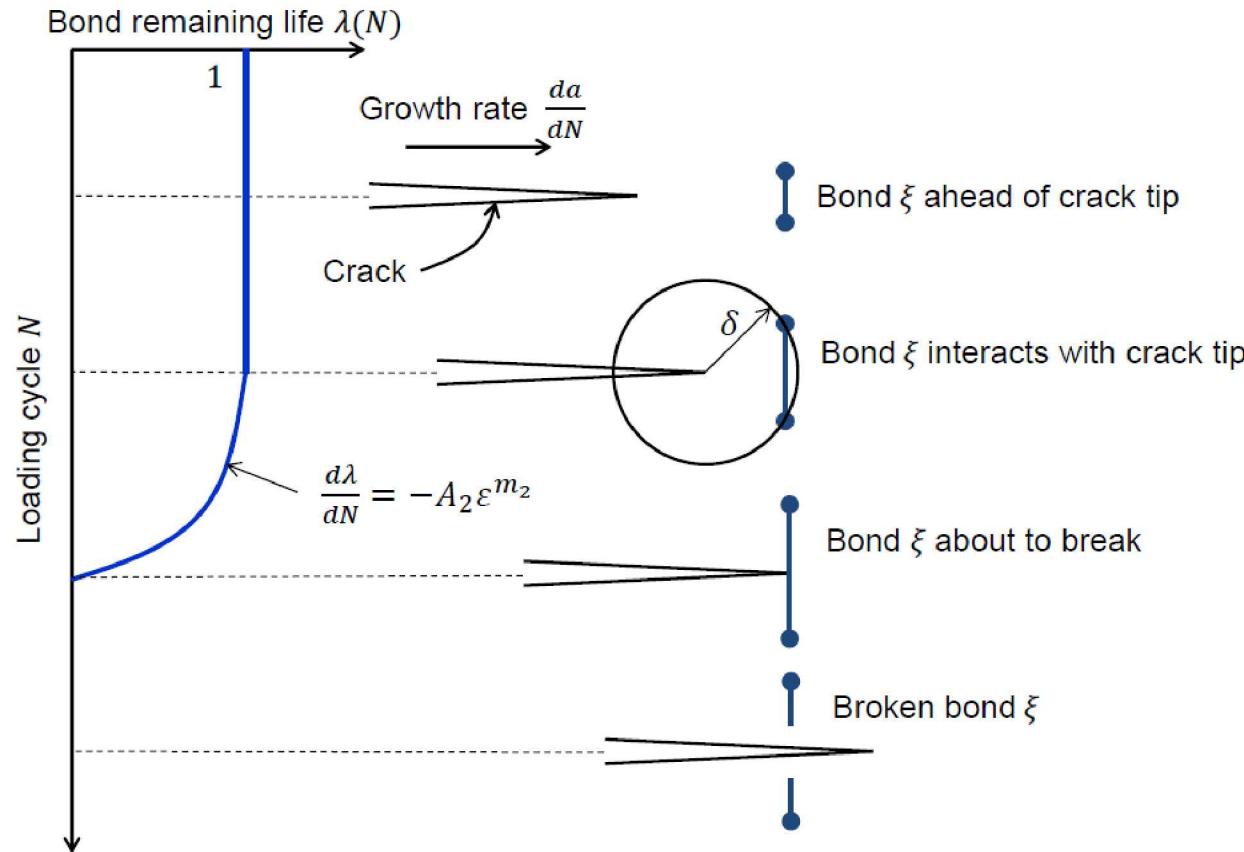
- Run many cyclic loading tests at different values of ε (constant for each test).
- For each test, compute when damage starts:

$$\frac{d\lambda}{dN}(N) = -A\varepsilon^m \quad \Rightarrow \quad N = \frac{1}{A\varepsilon^m}.$$

- Compare this to data on an ε - N plot, fit A and m .



Growth: Bonds interact with the strain field near an approaching crack



Relate crack growth to remaining life

- Evolution of remaining life:

$$\lambda(\delta) - \lambda(0) = \int_0^\delta \frac{d\lambda}{dz} dz = \int_0^\delta \frac{d\lambda}{dN} \frac{dN}{dz} dz.$$

- Recall

$$\frac{d\lambda}{dN} = -A\varepsilon^m.$$

- Denote by da/dN the crack growth rate.

$$1 - 0 = \frac{A}{da/dN} \int_0^\delta \varepsilon^m(z) dz$$

- Cyclic strain ahead of a crack:

$$\varepsilon(z) = \varepsilon_{\text{core}} f\left(\frac{z}{\delta}\right) = \frac{\Delta K}{E\sqrt{\delta}} f\left(\frac{z}{\delta}\right).$$

- Thus, for some c ,

$$\frac{da}{dN} = cA\Delta K^m$$

Phase II calibration from Paris Law data

- Now have

$$\frac{da}{dN} = cA\Delta K^m$$

where c and m are as yet unknown.

- Assume the Paris Law holds:

$$\frac{da}{dN} = C\Delta K^M$$

where C and M are constants that can be found from test data.

- Conclude

$$m = M.$$

- Need to do one computational simulation with an assumed value $C = 1$ to evaluate A .

Summary so far

- Each bond has a remaining life $\lambda(N)$:

$$\lambda(0) = 1, \quad \frac{d\lambda}{dN}(N) = -A\varepsilon^m, \quad \lambda \leq 0 \text{ means failure.}$$

- In Phase I, use A and m from S - N data.
- In Phase II, use a different calibration from Paris law data.

Time mapping permits very large N

- We can avoid modeling each cycle explicitly.
- Define the *loading ratio* by

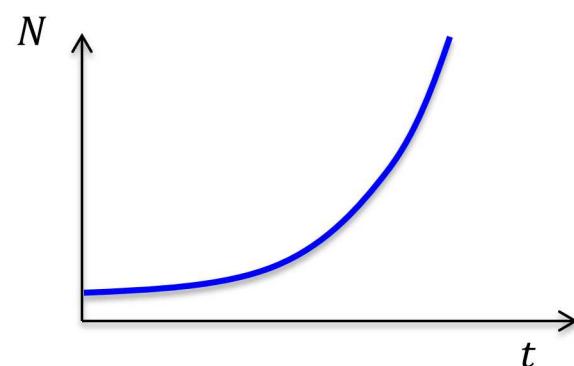
$$R = \frac{s^-}{s^+} \quad \Rightarrow \quad \varepsilon = |s^+ - s^-| = |(1 - R)s^+|.$$

- Map t to N :

$$N = e^{t/\tau}$$

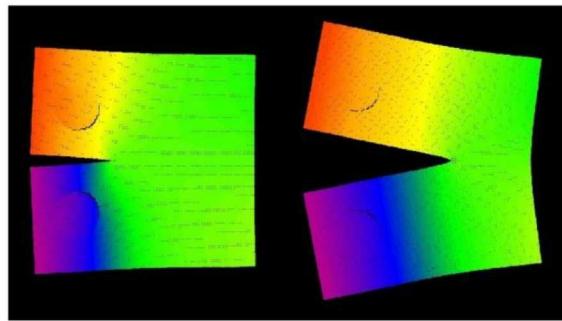
where τ is a constant chosen according to convenience.

- Fatigue model in terms of t instead of N :

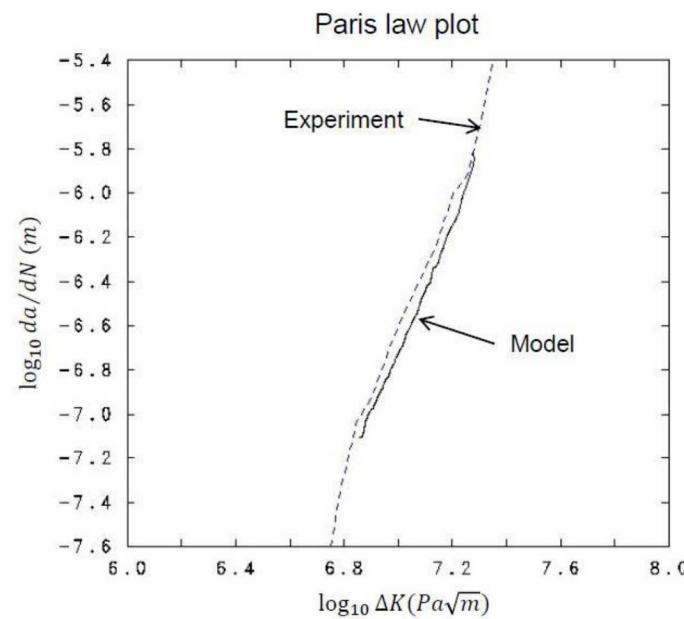
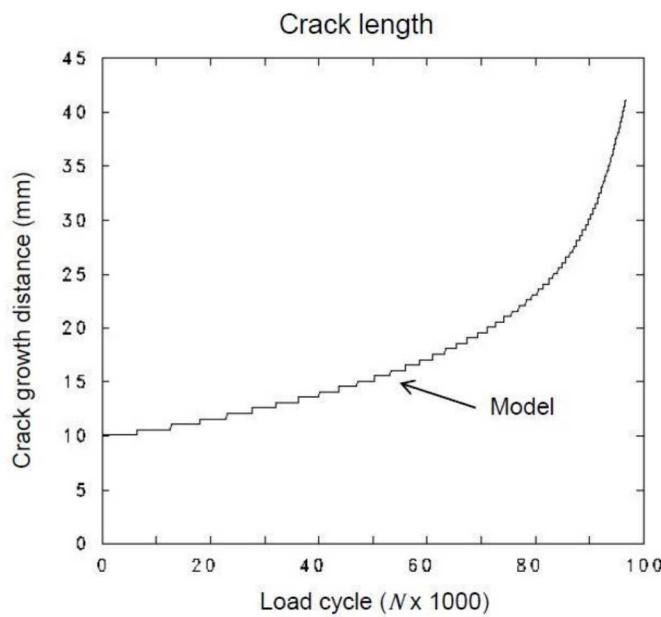


$$\frac{d\lambda}{dt} = \frac{d\lambda}{dN} \frac{dN}{dt} = \frac{-|1 - R|AN}{\tau} |s^+|^m.$$

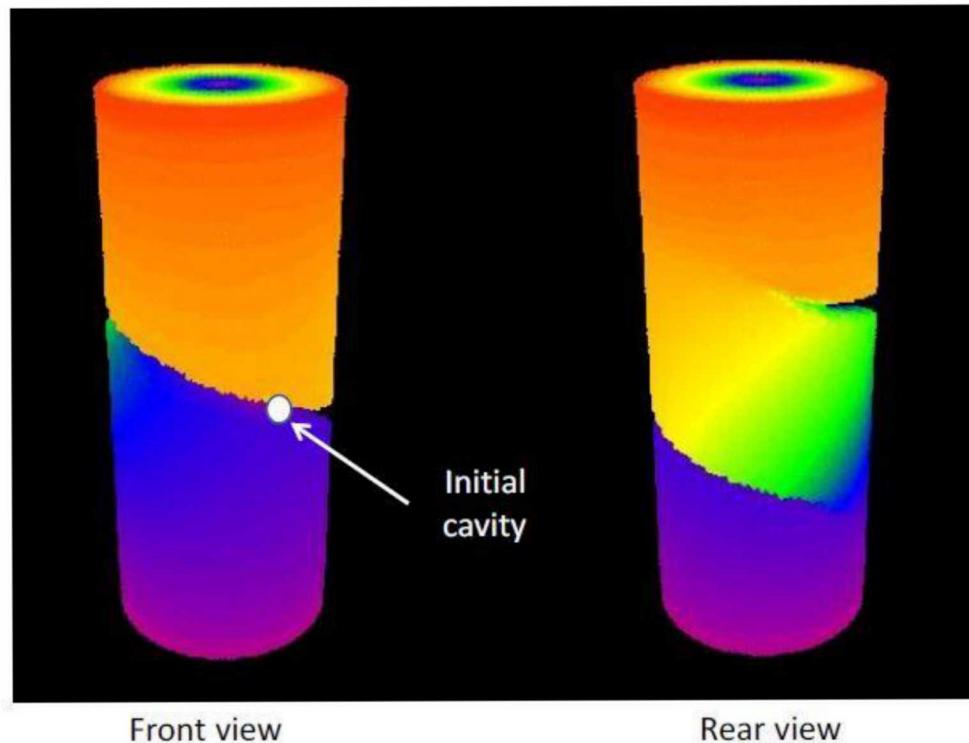
Aluminum compact test specimen



Data: T. Zhao and Y. Jiang. Fatigue of 7075-T651 aluminum alloy. International Journal of Fatigue, 30:834{849, 2008.



Spiral crack in a rod under torsion



Discussion

- Method retains the main advantages of peridynamics.
 - Autonomous crack growth
 - Includes both nucleation and growth phases
 - Permits interaction between multiple cracks
 - Arbitrary crack path in 3D.
- A simple enhancement allows a spectrum of loading frequencies
 - This is a peridynamic version of Miner's rule.