

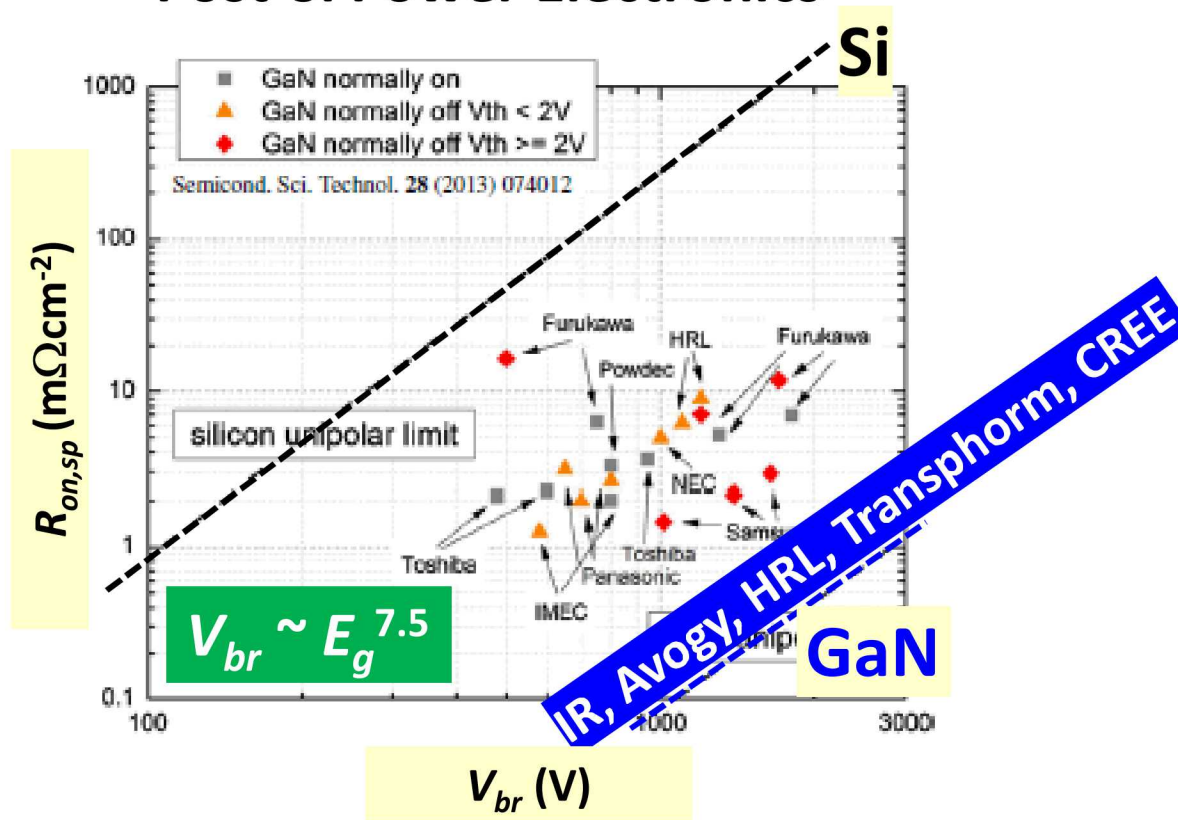
Strong sensitivity of Si doping efficiency and deep level formation on growth temperature for *n*-type $\text{Al}_{0.7}\text{Ga}_{0.3}\text{N}$

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Ultra-wide band gaps: Next frontier of power electronics

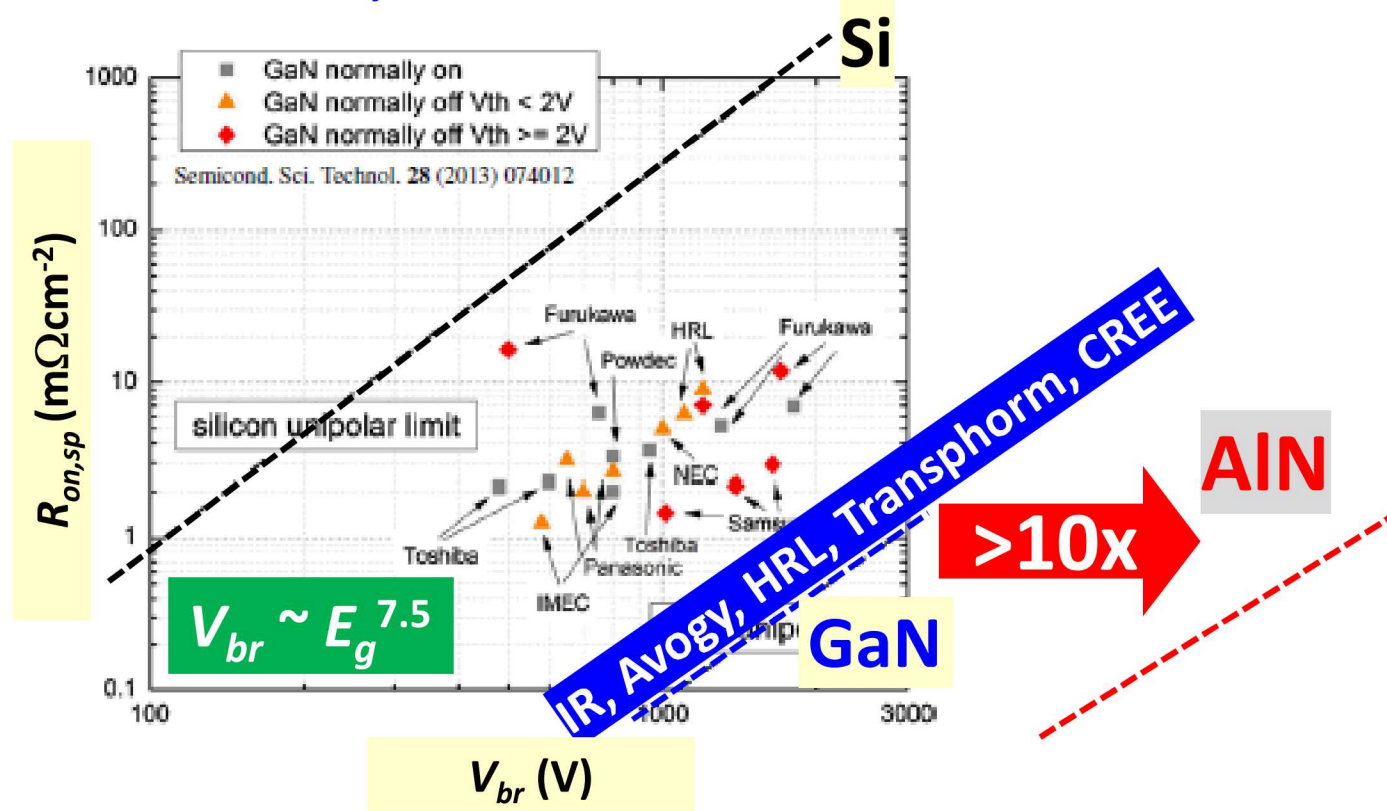
Post-Si Power Electronics



➤ Wide band gaps (WBGs) GaN and SiC displacing Si for power electronics

Ultra-wide band gaps: Next frontier of power electronics

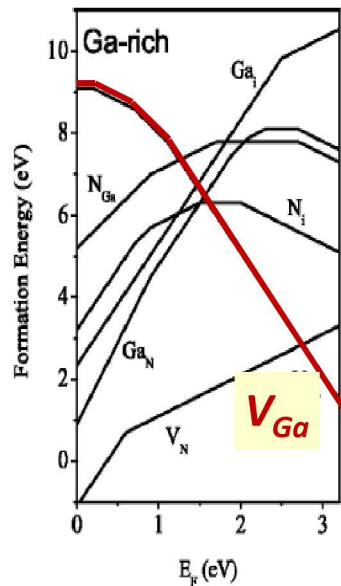
Post-SiC, Post-GaN Power Electronics



- Ultra-wide band gap (UWBG) semiconductors ($E_g > 4 \text{ eV}$) have potential for dramatic increase in breakdown voltage (V_{br})
- Controlling UWBG electrical conductivity → understanding fundamental relationship between UWBG doping and defects

Theory: UWBGs self-compensate by native defect formation

GaN defect formation

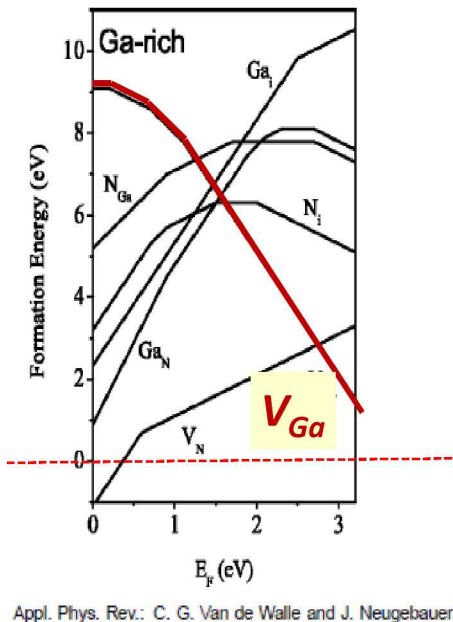


Appl. Phys. Rev.: C. G. Van de Walle and J. Neugebauer

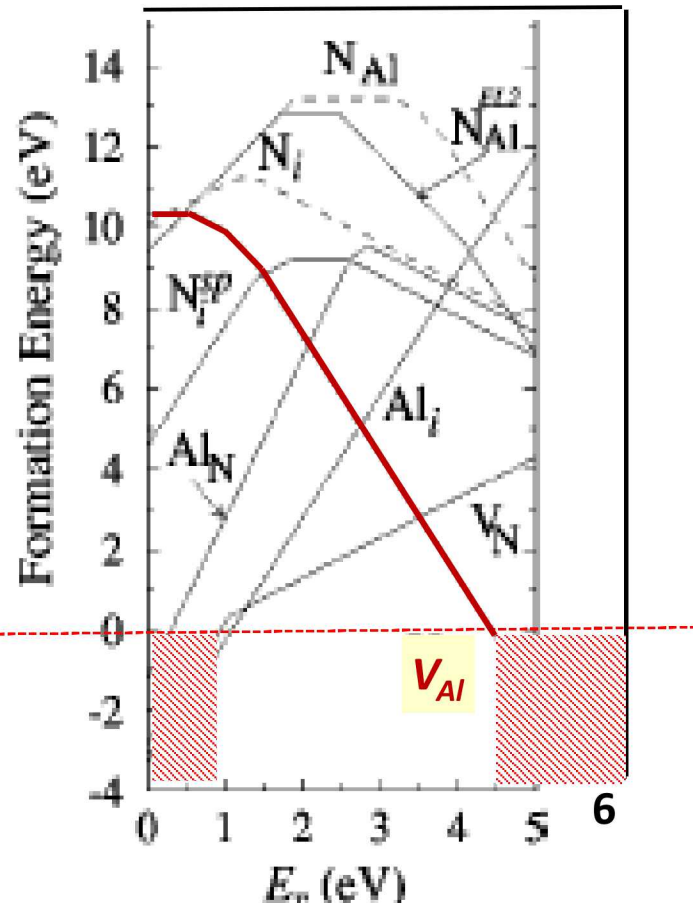
- Compensating defects become more favorable with increasing E_g
- Carrier capture reduces defect formation energy by $\sim E_g$

Theory: UWBGs self-compensate by vacancy formation

GaN defect formation



AlN defect formation



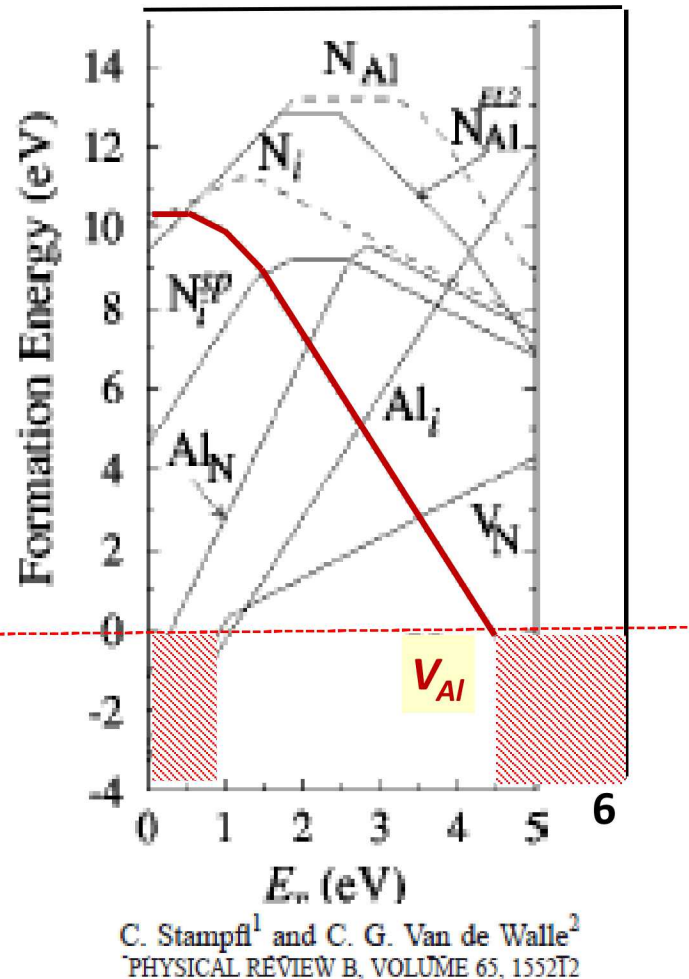
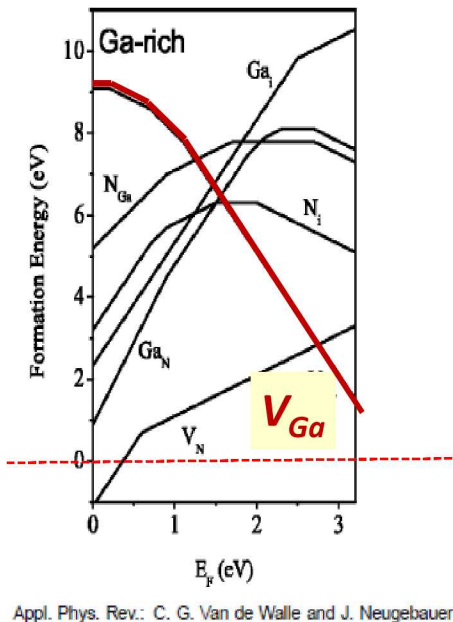
C. Stampfl¹ and C. G. Van de Walle²
¹PHYSICAL REVIEW B, VOLUME 65, 1552T2

- Vacancy formation energy becomes negligibly small for UWBGs
- Autonomic defect formation pins E_F and completely compensates dopants

Theory: UWBGs self-compensate by vacancy formation

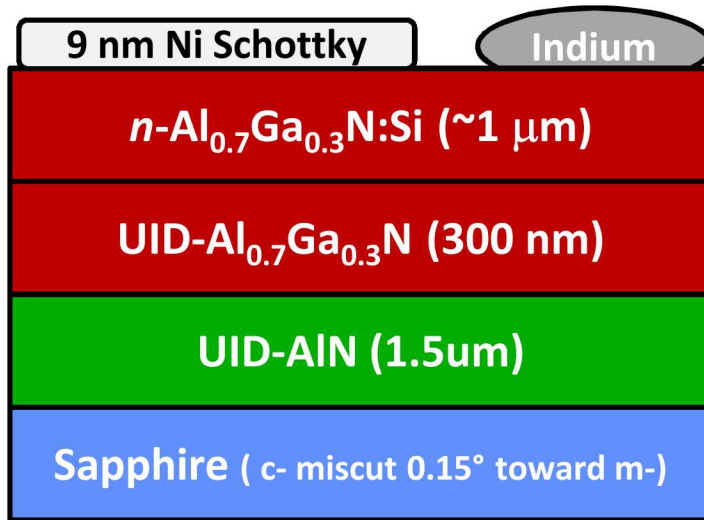
AlN defect formation

GaN defect formation



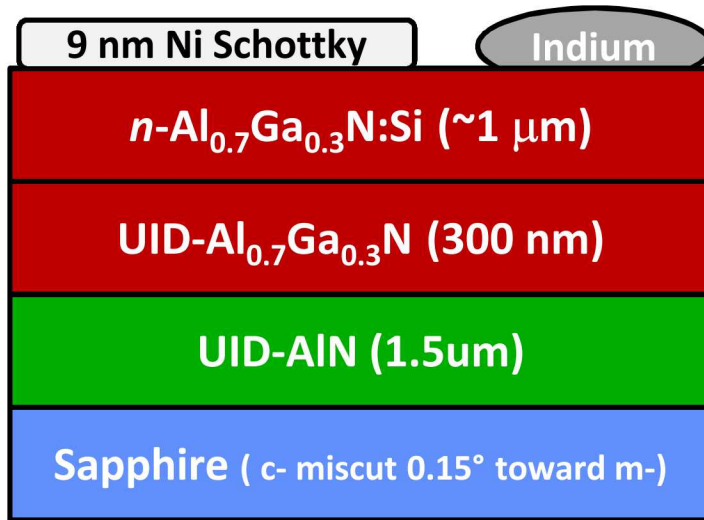
- Do compensating defects limit *n*-type doping in Al-rich AlGaN?
- If so, what are the physical origin of the defects?
- How can they be controlled?

Study interplay of defects and doping in $n\text{-Al}_{0.7}\text{Ga}_{0.3}\text{N}$



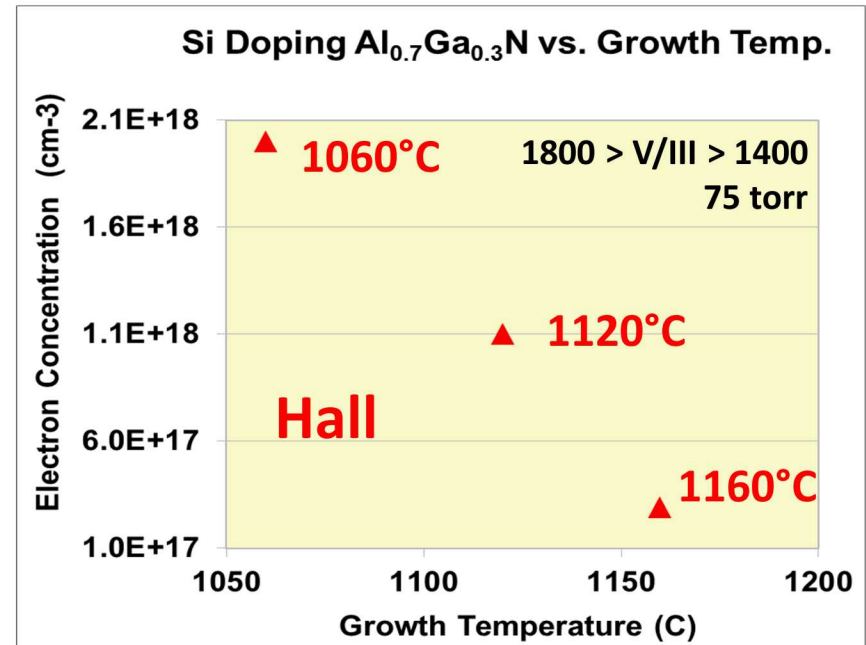
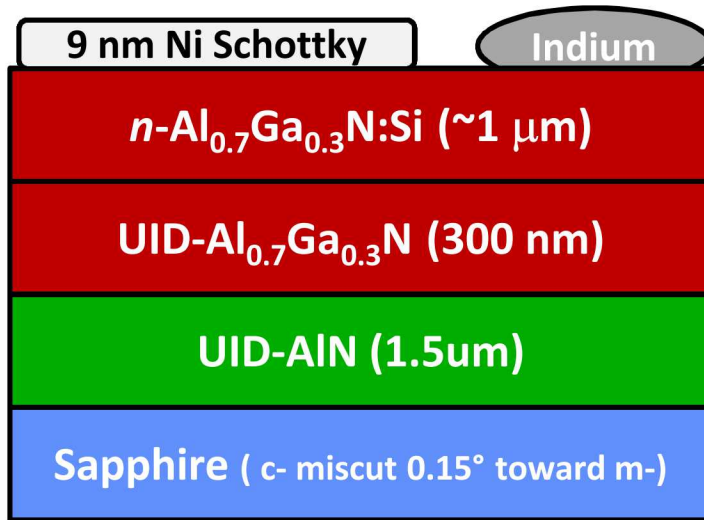
- $\text{Al}_{0.7}\text{Ga}_{0.3}\text{N}$ grown by metal-organic vapor phase epitaxy (MOVPE)
- TDD $\sim 1 - 3 \times 10^9\ \text{cm}^{-2}$

Study interplay of defects and doping in $n\text{-Al}_{0.7}\text{Ga}_{0.3}\text{N}$



- Growth temperature (T_g) range of 1060 - 1160 °C
 - Si flow rate constant
 - Increased TMGa flux (2x) to maintain $x_{\text{Al}} = 0.7$
 - Growth rate ($\mu\text{m/hr}$): 0.25 (1160 °C), 0.38 (1060 °C)
- Higher T_g improved surface morphology and likely reduced impurities¹

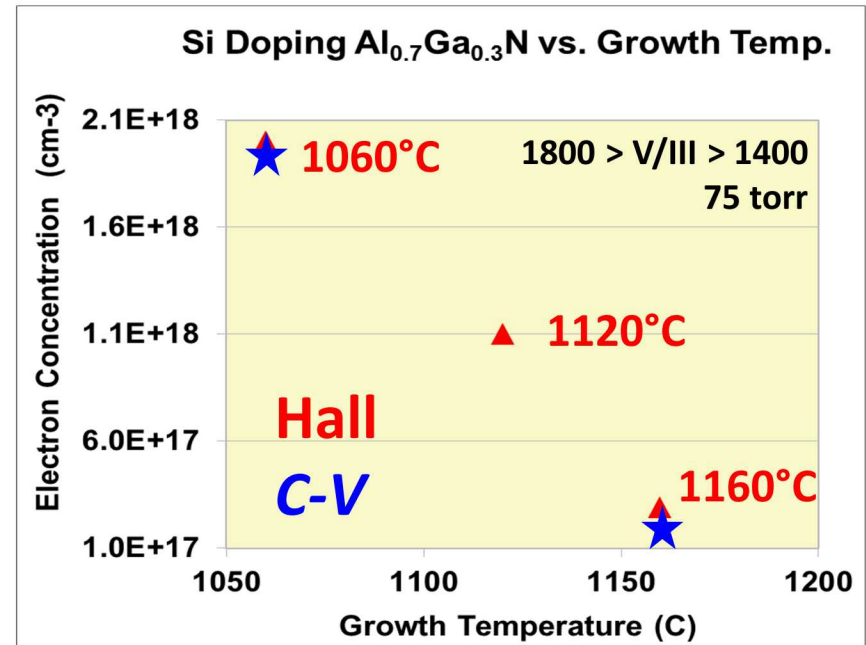
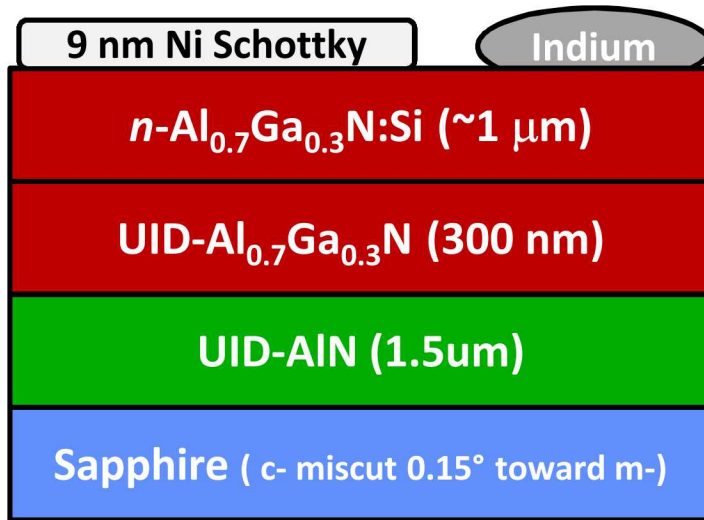
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➤ $\text{Al}_{0.7}\text{Ga}_{0.3}\text{N}$ dopant efficacy 20x lower with higher T_g ¹

Study interplay of defects and doping in $n\text{-Al}_{0.7}\text{Ga}_{0.3}\text{N}$



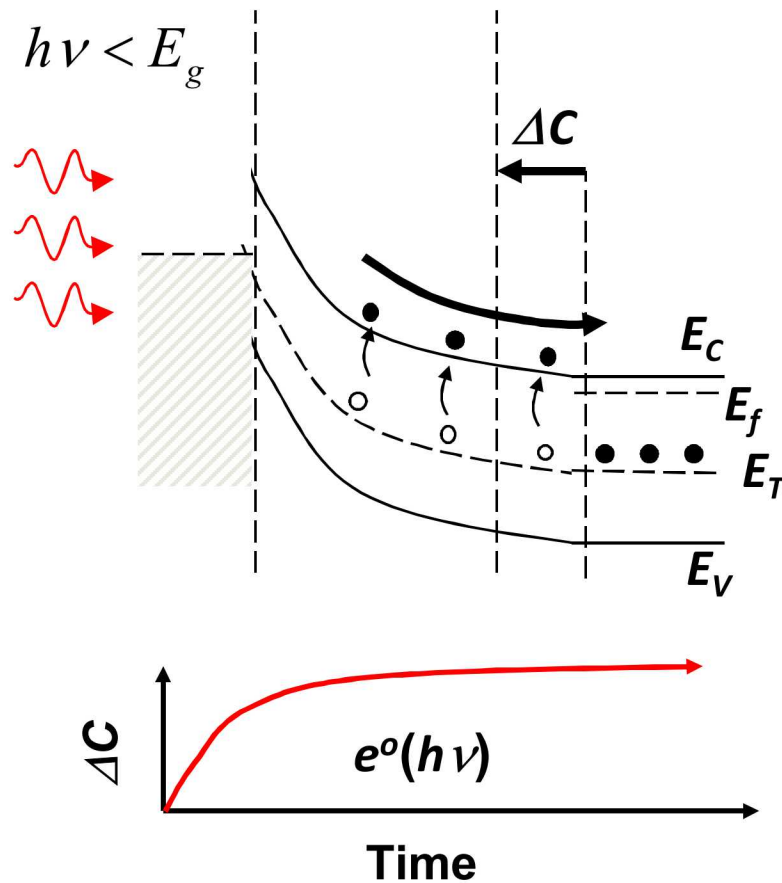
- C-V and Hall show Si dopant remained shallow
- Decrease in n due to deep level defect compensation

Use deep level optical spectroscopy (DLOS) to understand how T_g mediates defects and doping in $n\text{-Al}_{0.7}\text{Ga}_{0.3}\text{N}$

Deep Level Optical Spectroscopy

Deep Level Optical Spectroscopy (DLOS)¹

- Photocapacitance technique
- Electrical measurement of optical absorption by deep level defects
- Sub-band gap optical stimulation to photoionize defect levels

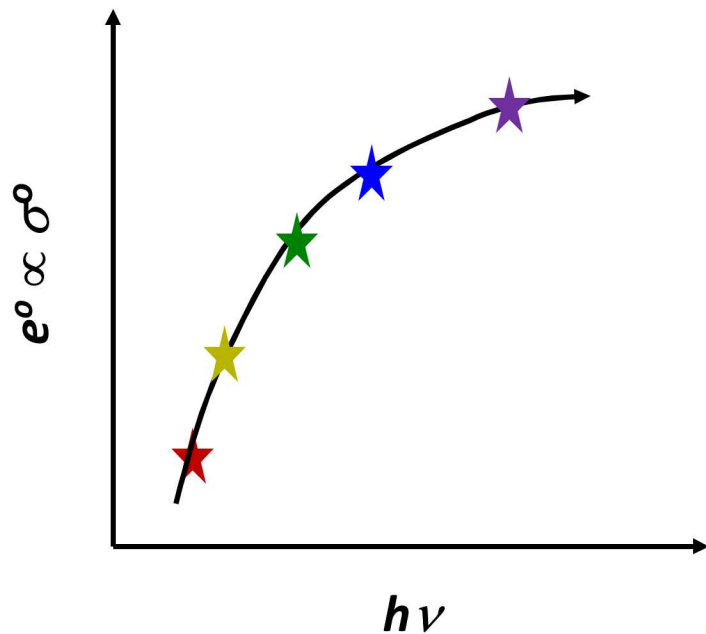


Deep Level Optical Spectroscopy

Deep Level Optical Spectroscopy (DLOS)¹

- Fundamental physics from optical cross-section: $\sigma^o(h\nu) = e^o(h\nu)/\phi(h\nu) = \alpha/N_t$
- Determine deep level energy E_o from lineshape of $\sigma^o(h\nu)$
- Discuss defect density later...

Optical analog of Arrhenius plot

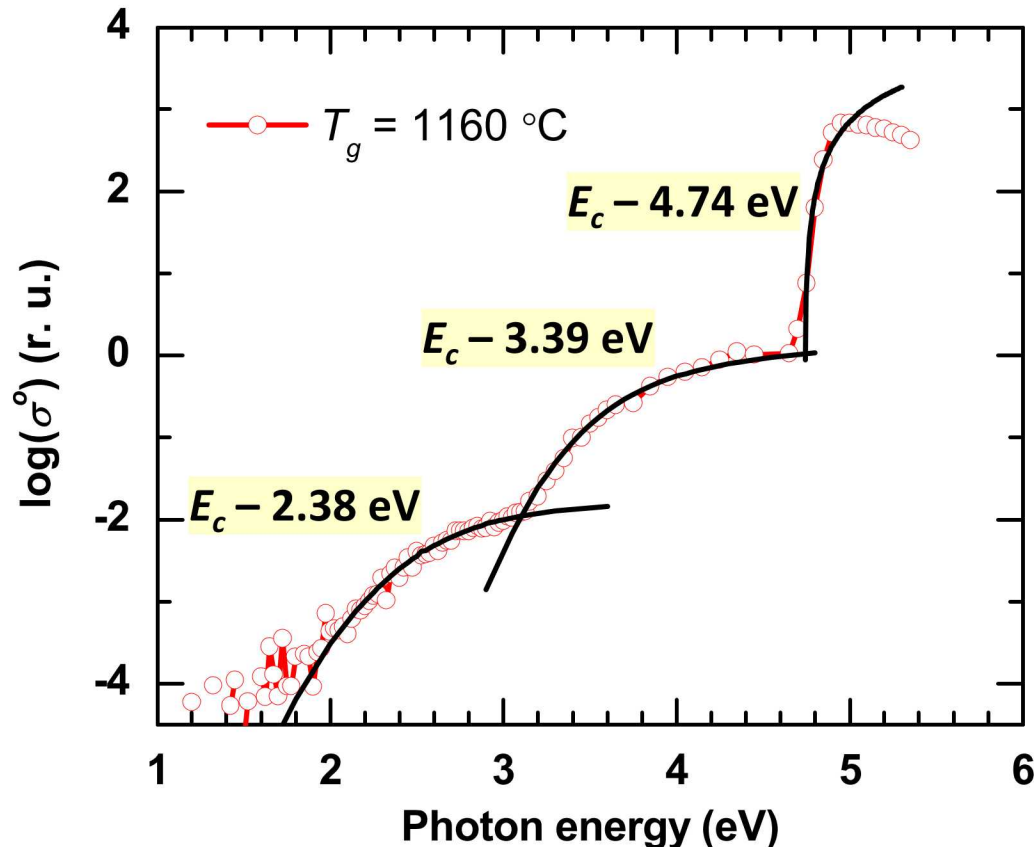


$$\sigma^o(h\nu) \propto \frac{1}{E_o} \left\{ \left(\frac{E_o}{h\nu} \right) \cdot \left(1 - \frac{E_o}{h\nu} \right) \right\}^{3/2}$$

$$\sigma^o(h\nu, T) \propto \frac{E_o^2}{h\nu \sqrt{4k_B T \pi d_{FC}}} \int_1^\infty dx \sqrt{\frac{x-1}{E_o}} \frac{1}{[(x-1)/E_o + m]^2} \times \exp \left[-\frac{E_o^2 (h\nu/E_o - x)^2}{4k_B T d_{FC}} \right], \quad (4)$$

DLOS identifies $\text{Al}_{0.7}\text{Ga}_{0.3}\text{N}$ deep levels

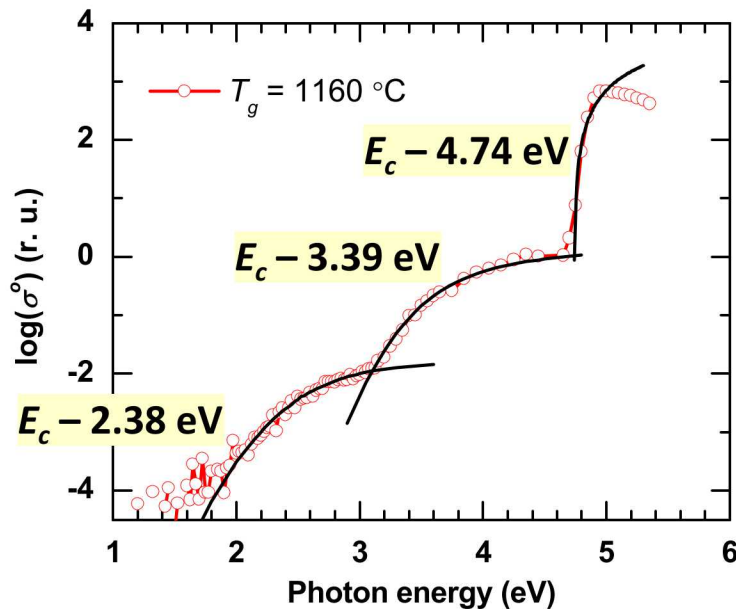
DLOS of $n\text{-Al}_{0.7}\text{Ga}_{0.3}\text{N}$ for $T_g = 1160^\circ\text{C}$



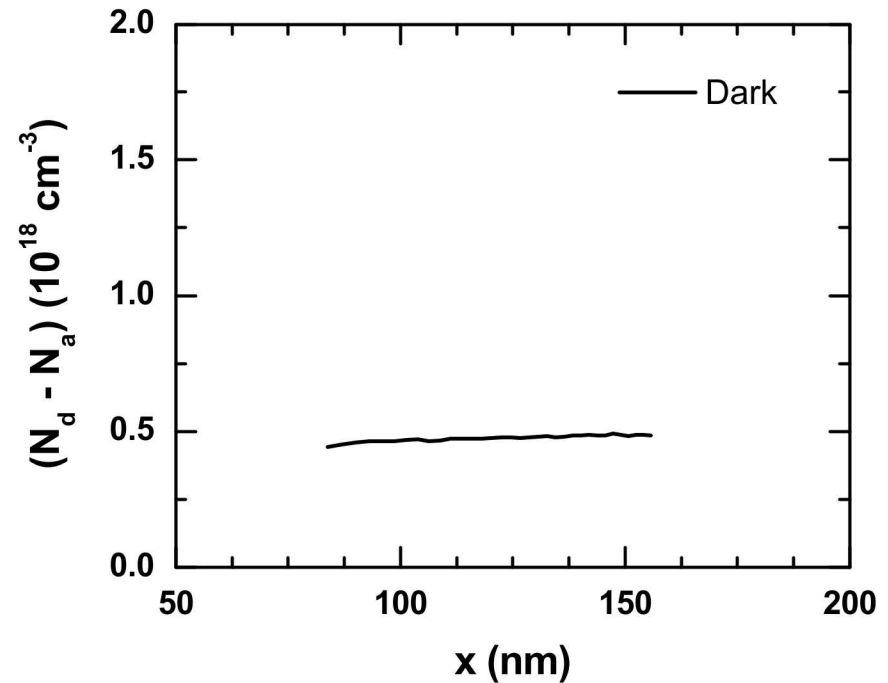
- Observe three deep level defects
- DLOS spectra does not convey defect density information

Lighted Capacitance-Voltage identifies compensators

DLOS of $n\text{-Al}_{0.7}\text{Ga}_{0.3}\text{N}$ for $T_g = 1160^\circ\text{C}$



LCV of $n\text{-Al}_{0.7}\text{Ga}_{0.3}\text{N}$ for $T_g = 1160^\circ\text{C}$

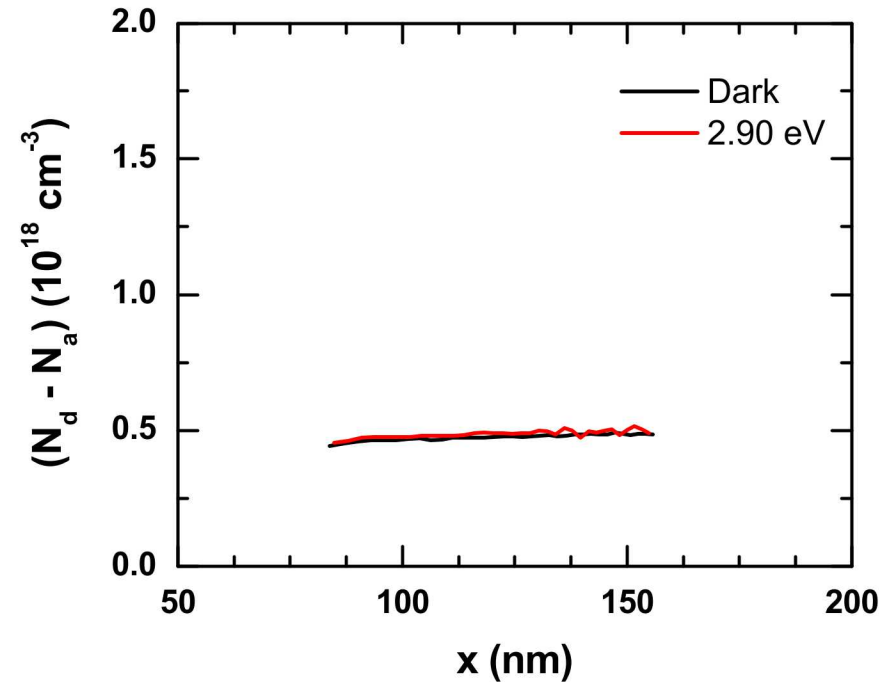
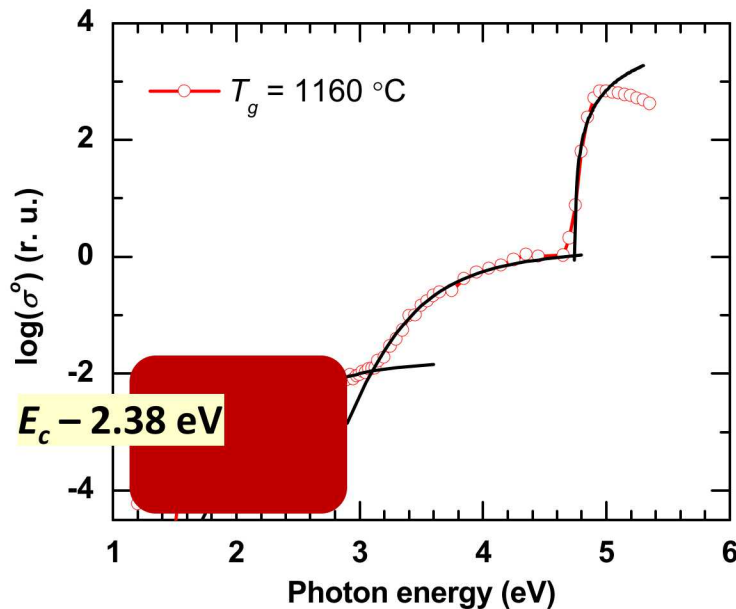


- Lighted Capacitance-Voltage (LCV) measures defect density (N_t)
- $N_t = \Delta(N_d - N_a)$ measured by CV under sub-gap, monochromatic illumination

Lighted Capacitance-Voltage identifies compensators

LCV for VNA4153A

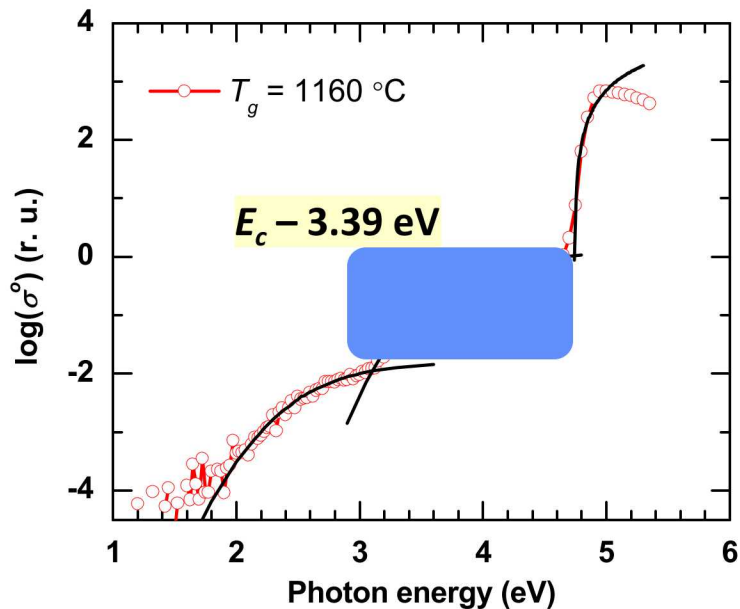
DLOS of $n\text{-Al}_{0.7}\text{Ga}_{0.3}\text{N}$ for $T_g = 1160^\circ\text{C}$



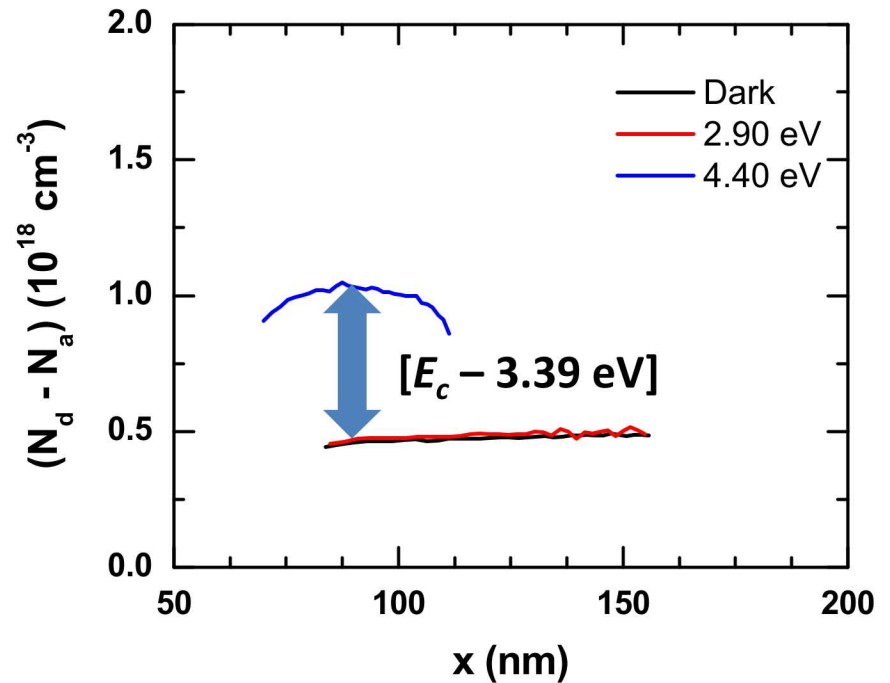
➤ $E_c - 2.38 \text{ eV}$ level is not a strong compensation center

Lighted Capacitance-Voltage identifies compensators

DLOS of $n\text{-Al}_{0.7}\text{Ga}_{0.3}\text{N}$ for $T_g = 1160\text{ }^\circ\text{C}$



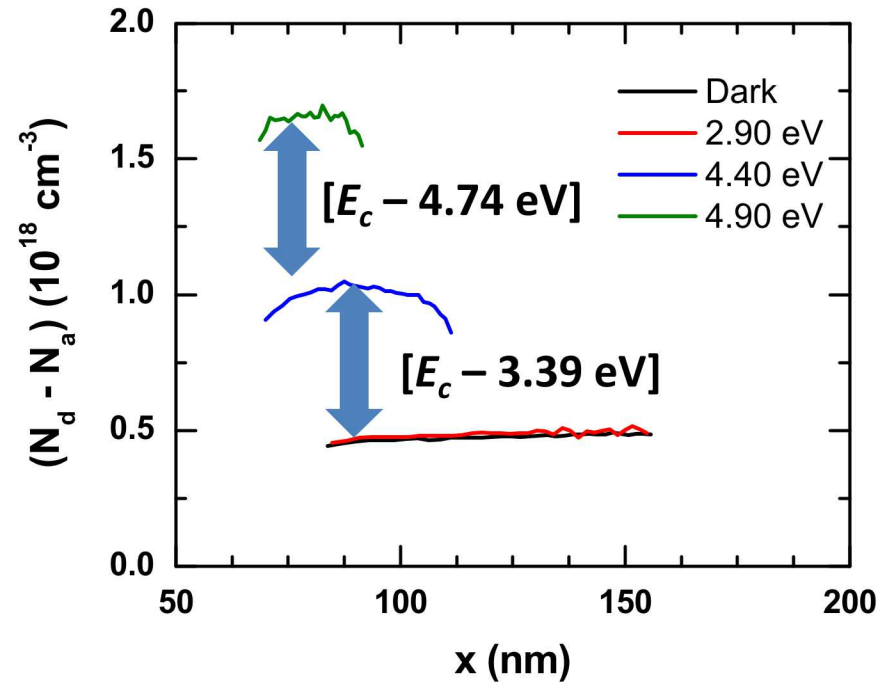
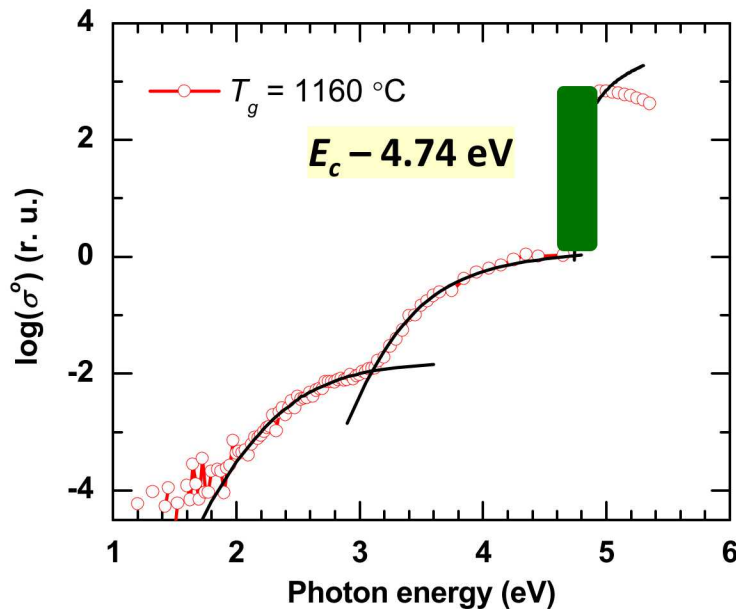
LCV for VNA4153A



Lighted Capacitance-Voltage identifies compensators

LCV for VNA4153A

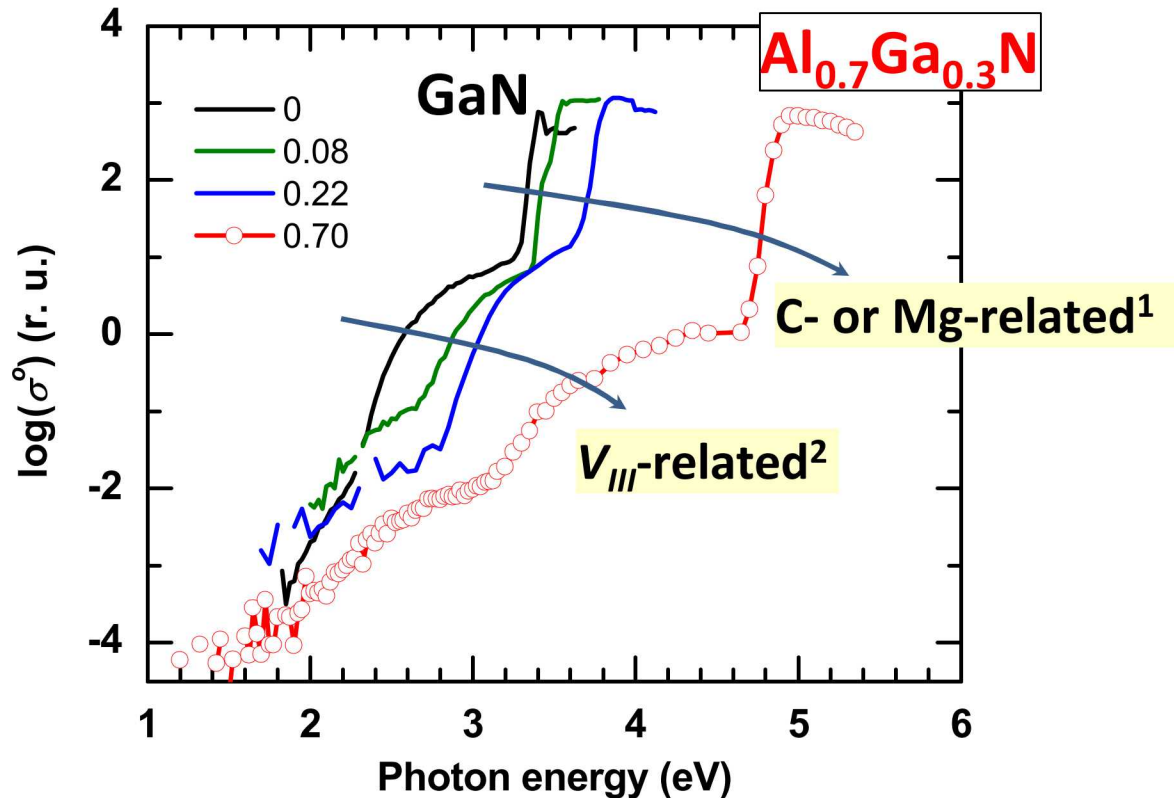
DLOS of $n\text{-Al}_{0.7}\text{Ga}_{0.3}\text{N}$ for $T_g = 1160\text{ }^\circ\text{C}$



➤ $E_c - 3.39\text{ eV}$ and $E_c - 4.74\text{ eV}$ defects are dominate compensating centers

Identify physical origin from DLOS alloying study

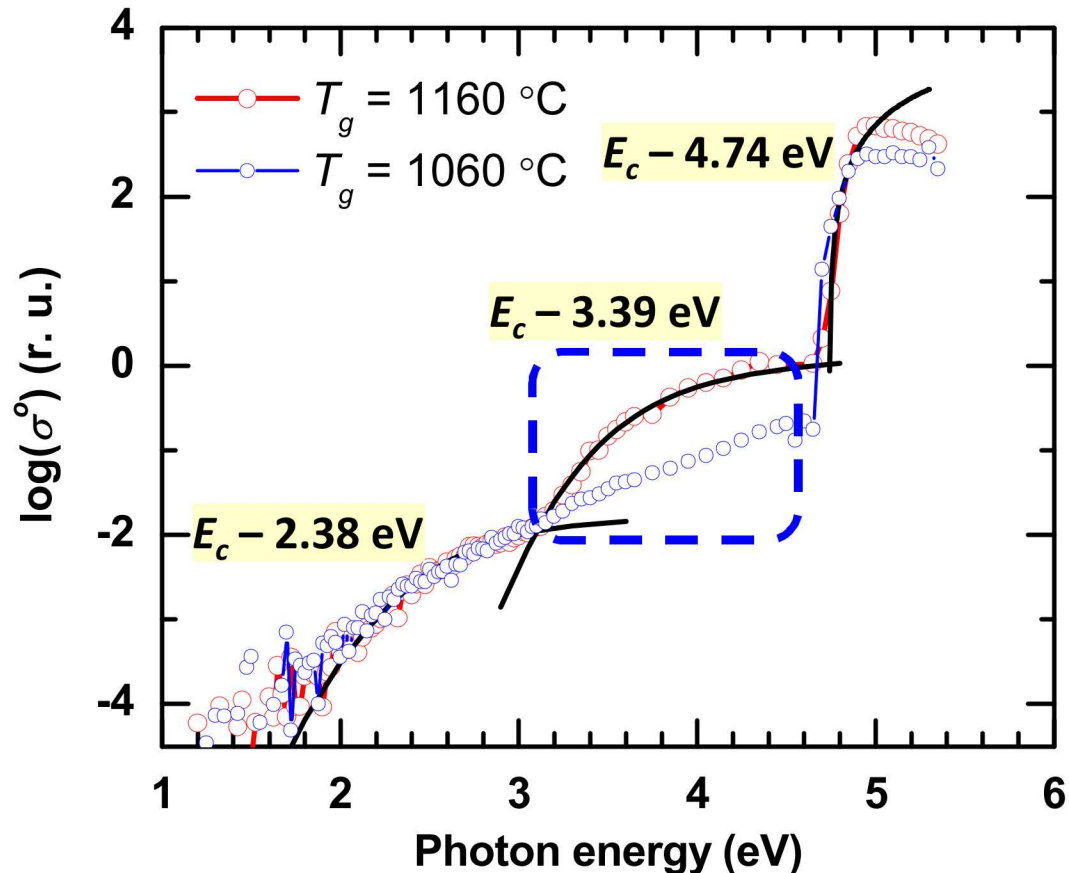
DLOS of (Al)GaN vs. Al mole fraction



- Systematic evolution suggests similar defect origins for GaN & $\text{Al}_{0.7}\text{Ga}_{0.3}\text{N}$
- Attribute $E_c - 4.74$ eV level to C or Mg impurities¹
- Attribute $E_c - 3.39$ eV level to V_{III} -related defect²

Lower $\text{Al}_{0.7}\text{Ga}_{0.3}\text{N}$ T_g suppresses defect formation

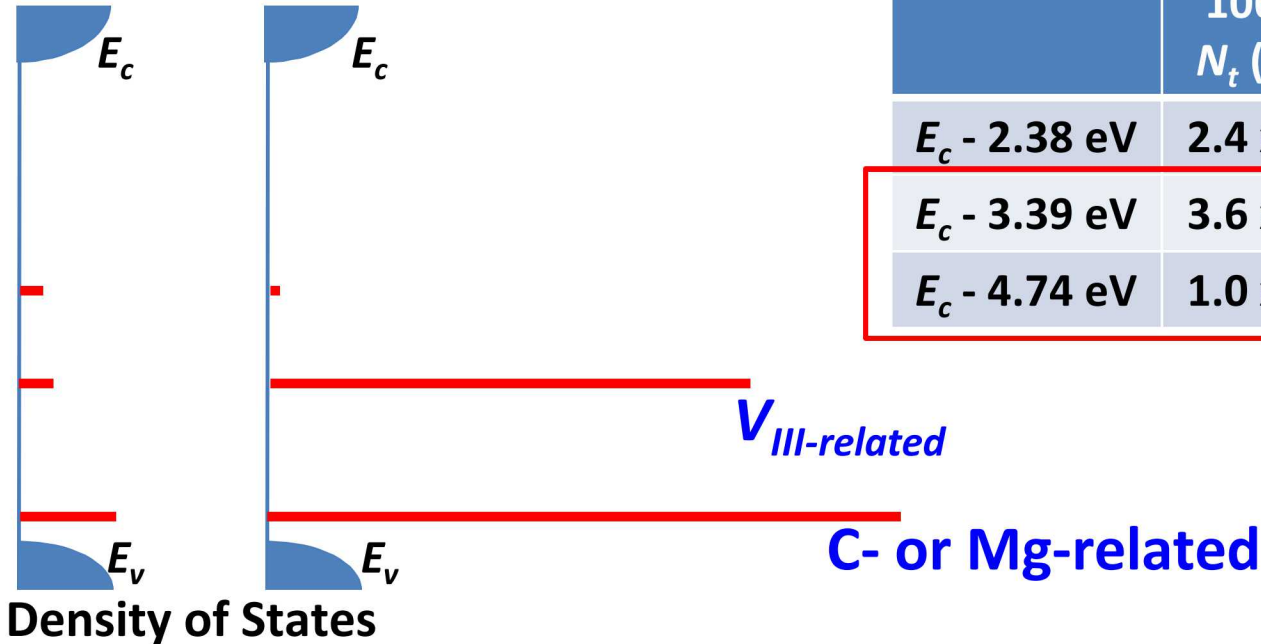
$n\text{-Al}_{0.7}\text{Ga}_{0.3}\text{N}$ DLOS spectra vs. T_g



➤ V_{III} -related $E_c - 3.39\text{ eV}$ deep level defect weakly evident for $T_g = 1060^\circ\text{C}$

Lower $\text{Al}_{0.7}\text{Ga}_{0.3}\text{N}$ T_g suppresses defect formation

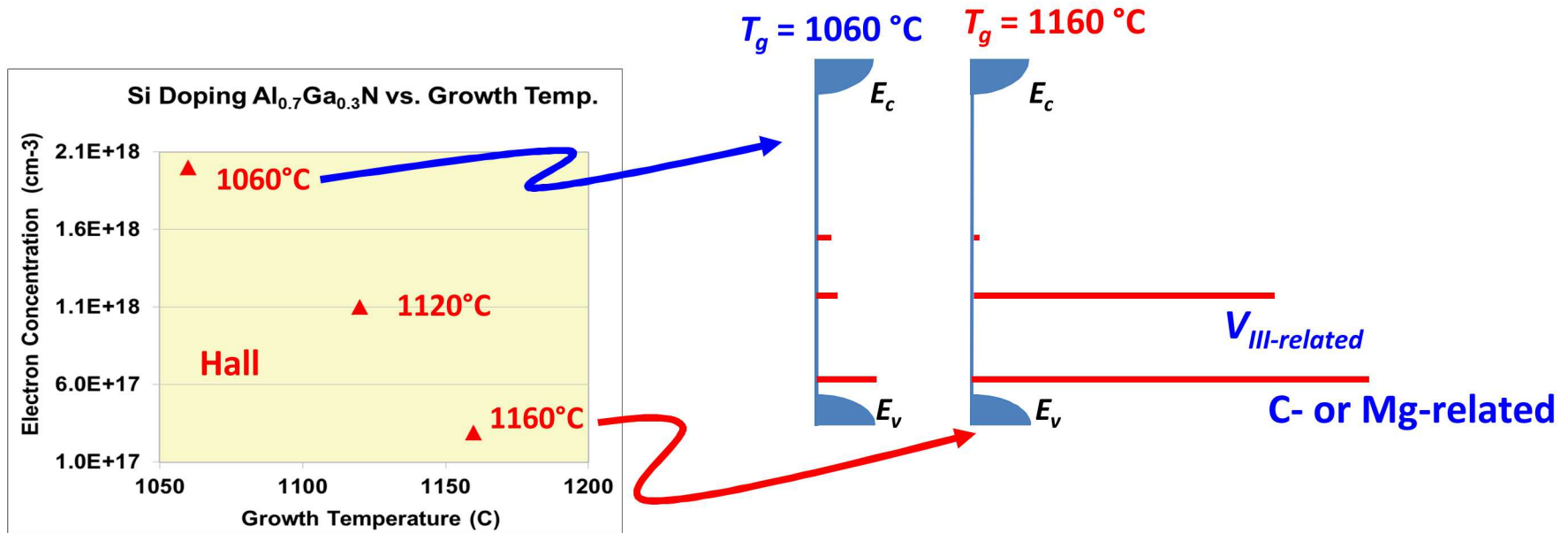
$T_g = 1060\text{ }^\circ\text{C}$ $T_g = 1160\text{ }^\circ\text{C}$



	1060 °C N_t (cm ⁻³)	1160 °C N_t (cm ⁻³)
$E_c - 2.38$ eV	2.4×10^{16}	1×10^{16}
$E_c - 3.39$ eV	3.6×10^{16}	5.1×10^{17}
$E_c - 4.74$ eV	1.0×10^{17}	6.6×10^{17}

➤ Net 10x reduction in V_{III} and impurity defects with lower $T_g \rightarrow$ **Unexpected**

Summary and Conclusions



- Si dopant compensation strongly dependent on $\text{Al}_{0.7}\text{Ga}_{0.3}\text{N}$ T_g
- Lower T_g reduced compensating V_{III} and impurity defects by 10x
- Lowering T_g may inhibit the very strong thermodynamic drive to form compensating point defects in UWBGs like Al-rich AlGaN
- Maintaining electrical quality of $\text{Al}_x\text{Ga}_{1-x}\text{N}$ ($x > 0.7$) at $T_g > 1100^\circ\text{C}$ could require intricate balance of growth conditions