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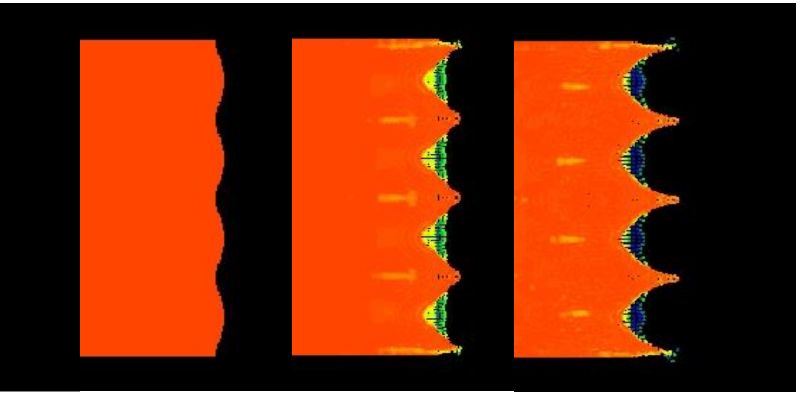

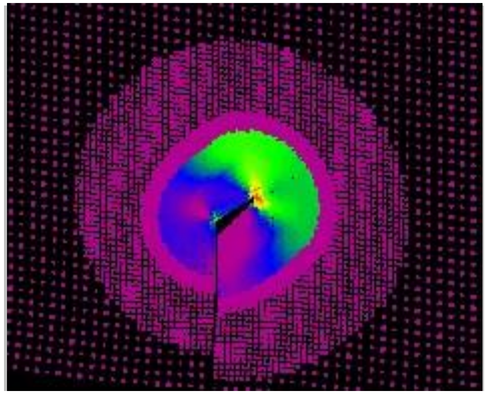

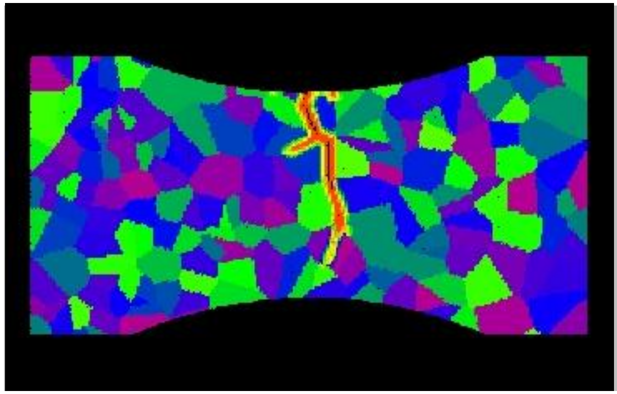

Unifying the mechanics of continua, cracks, and particles

Stewart Silling

Sandia National Laboratories

Albuquerque, New Mexico

Mechanical Engineering Seminar, Wayne State University, October 17, 2014

U.S. DEPARTMENT OF ENERGY

NNSA National Nuclear Security Administration

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Supercomputing when I was a student (~1972)

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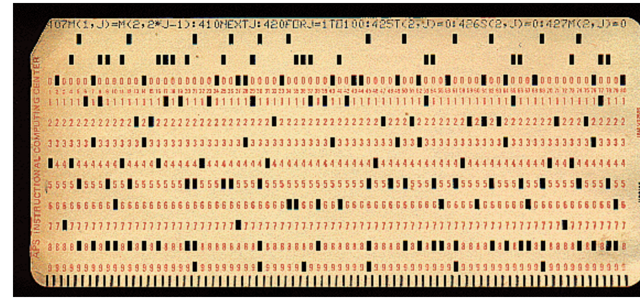


Punchcard

Control console, disk and tape drives

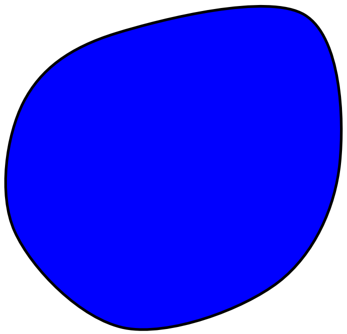
CDC 6600: 10MHz

Outline

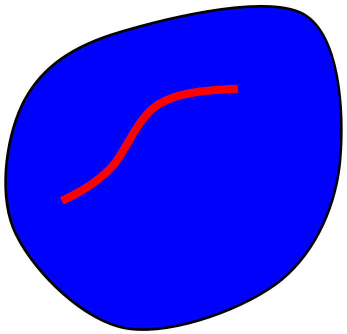
- Purpose of peridynamics
- Basic equations
- Dynamic fracture examples
- Continuum-particle connection: self-assembly
- Nonlocality in heterogeneous media: composites
- Multiscale peridynamics

Purpose of peridynamics

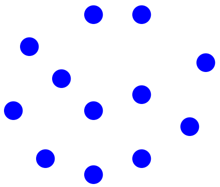
- To unify the mechanics of continuous and discontinuous media within a single, consistent set of equations.



Continuous body



Continuous body with a defect

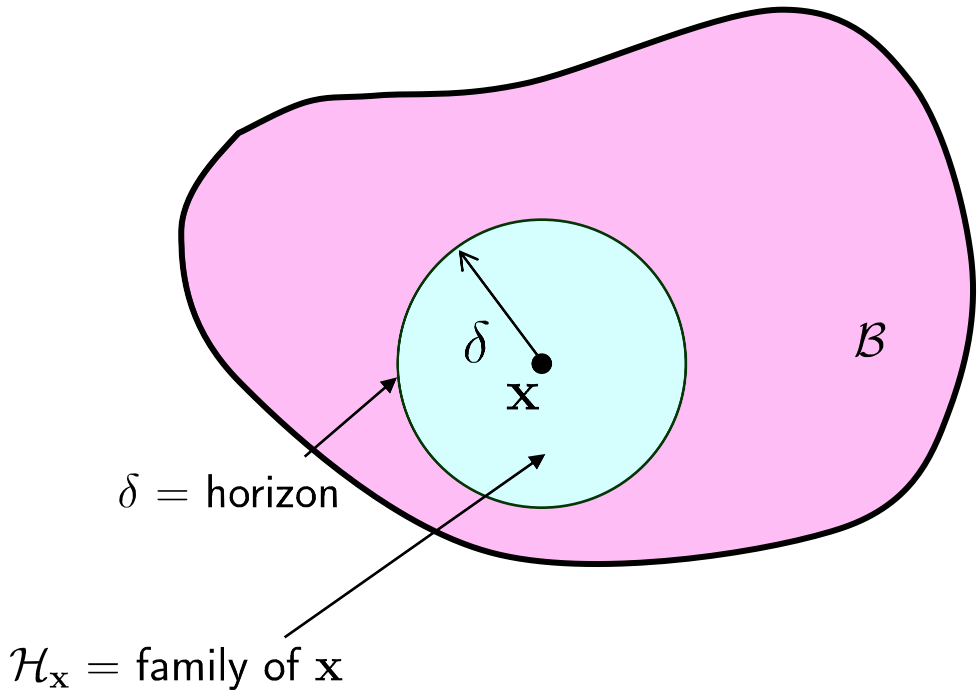


Discrete particles

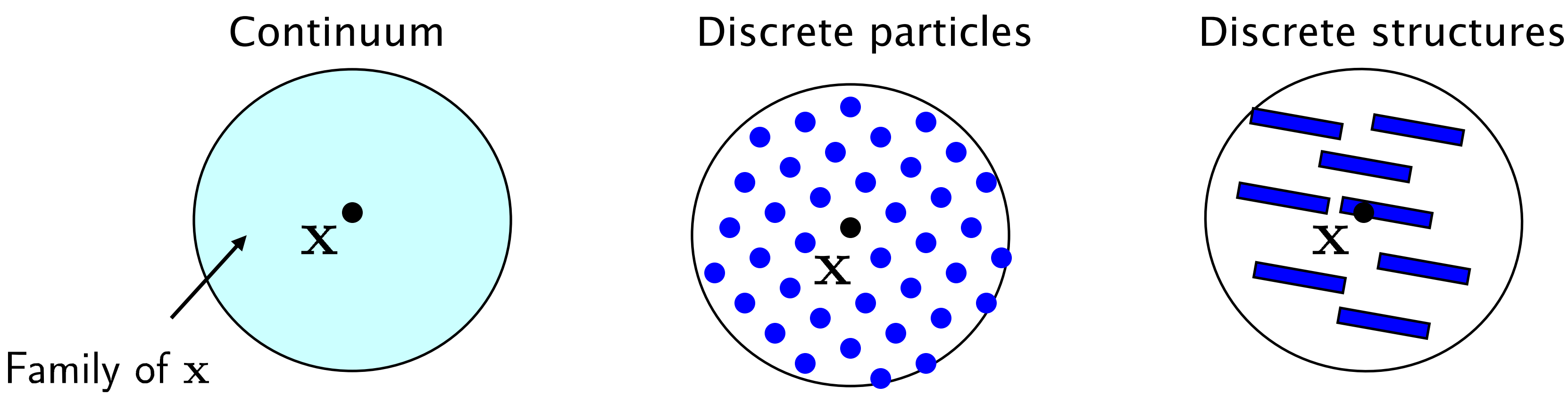
- Why do this?
  - Avoid coupling dissimilar mathematical systems (A to C).
  - Model complex fracture patterns.
  - Communicate across length scales.

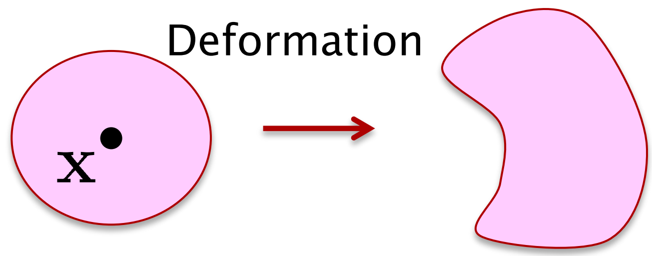
Peridynamics basics: Horizon and family

- Any point $\mathbf{x}$ interacts directly with other points within a distance δ called the "horizon."
- The material within a distance δ of $\mathbf{x}$ is called the "family" of $\mathbf{x}$, $\mathcal{H}_\mathbf{x}$.



Strain energy at a point





- Key assumption: the strain energy density at $\mathbf{x}$ is determined by the deformation of its family.

Potential energy minimization yields the peridynamic equilibrium equation

- Potential energy:

$$\Phi = \int_{\mathcal{B}} (W - \mathbf{b} \cdot \mathbf{y}) \, dV_{\mathbf{x}}$$

where W is the strain energy density, $\mathbf{y}$ is the deformation map, $\mathbf{b}$ is the applied external force density, and $\mathcal{B}$ is the body.

- Euler-Lagrange equation is the equilibrium equation:

$$\int_{\mathcal{H}_{\mathbf{x}}} \mathbf{f}(\mathbf{q}, \mathbf{x}) \, dV_{\mathbf{q}} + \mathbf{b}(\mathbf{x}) = 0$$

for all $\mathbf{x}$.

Peridynamics basics:
Bonds and bond force density

- The vector from $\mathbf{x}$ to any point $\mathbf{q}$ in its family in the reference configuration is called a *bond*.

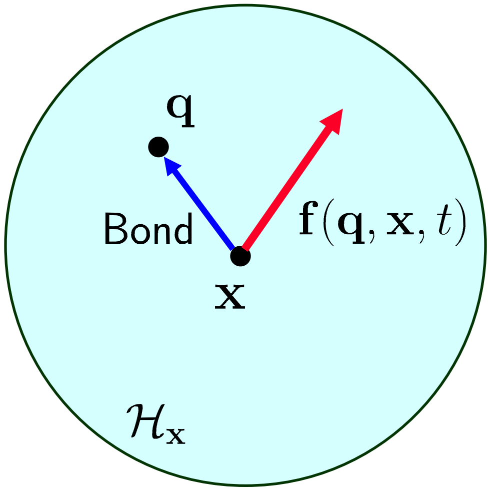
$$\boldsymbol{\xi} = \mathbf{q} - \mathbf{x}$$

- Each bond has a *pairwise force density* vector that is applied at both points:

$$\mathbf{f}(\mathbf{q}, \mathbf{x}, t).$$

- Equation of motion is an integro-differential equation, not a PDE:

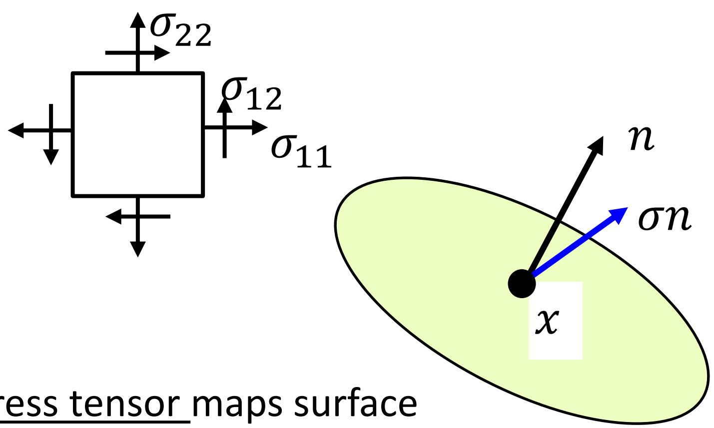
$$\rho(\mathbf{x})\ddot{\mathbf{y}}(\mathbf{x}, t) = \int_{\mathcal{H}} \mathbf{f}(\mathbf{q}, \mathbf{x}, t)\, dV_{\mathbf{q}} + \mathbf{b}(\mathbf{x}, t).$$



Peridynamics basics: The nature of internal forces

Standard theory

Stress tensor field
(assumes continuity of forces)



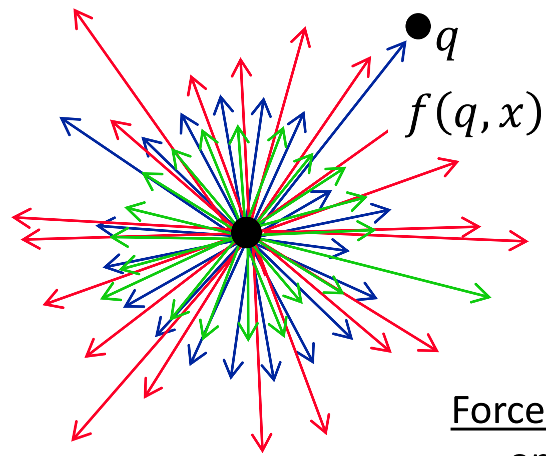
Stress tensor maps surface normal vectors onto surface forces

$$\rho\ddot{u}(x,t) = \nabla \cdot \sigma(x,t) + b(x,t)$$

Differentiation of surface forces

Peridynamics

Bond forces between neighboring points
(allowing discontinuity)



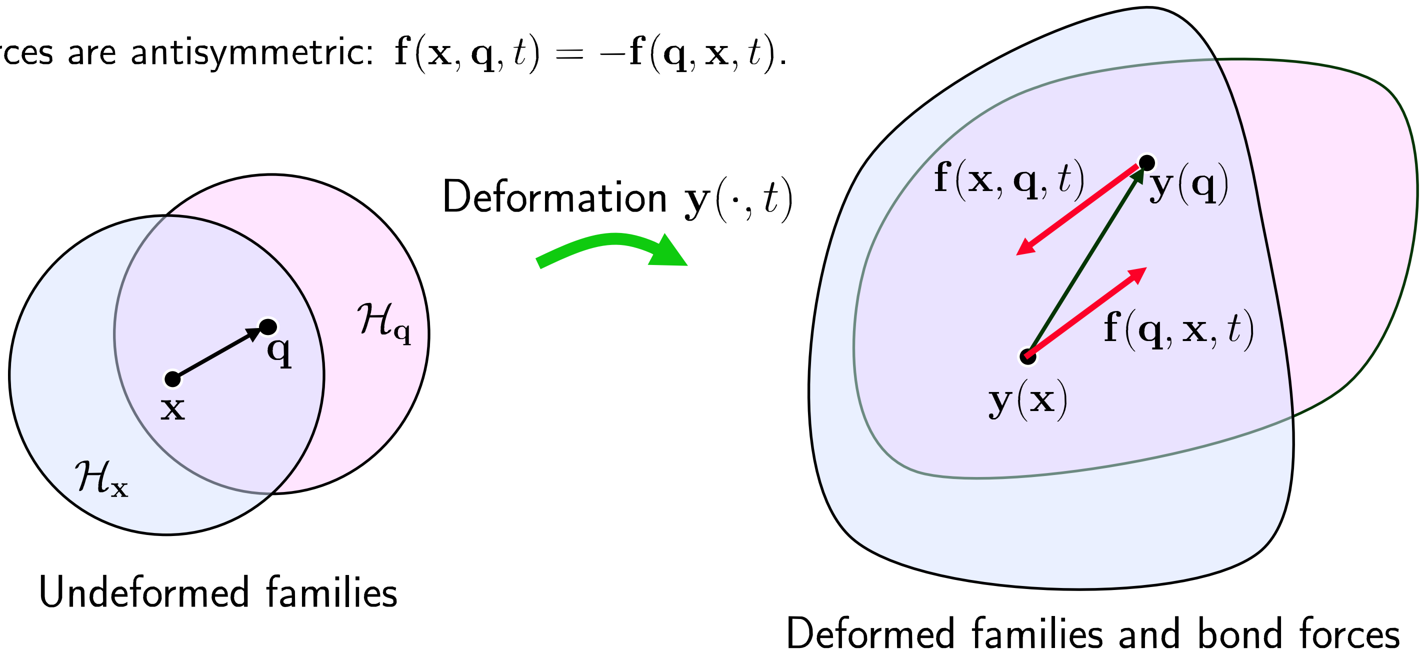
Force state maps bonds onto bond forces

$$\rho\ddot{u}(x,t) = \int_{H_x} f(q,x)dV_q + b(x,t)$$

Summation over bond forces

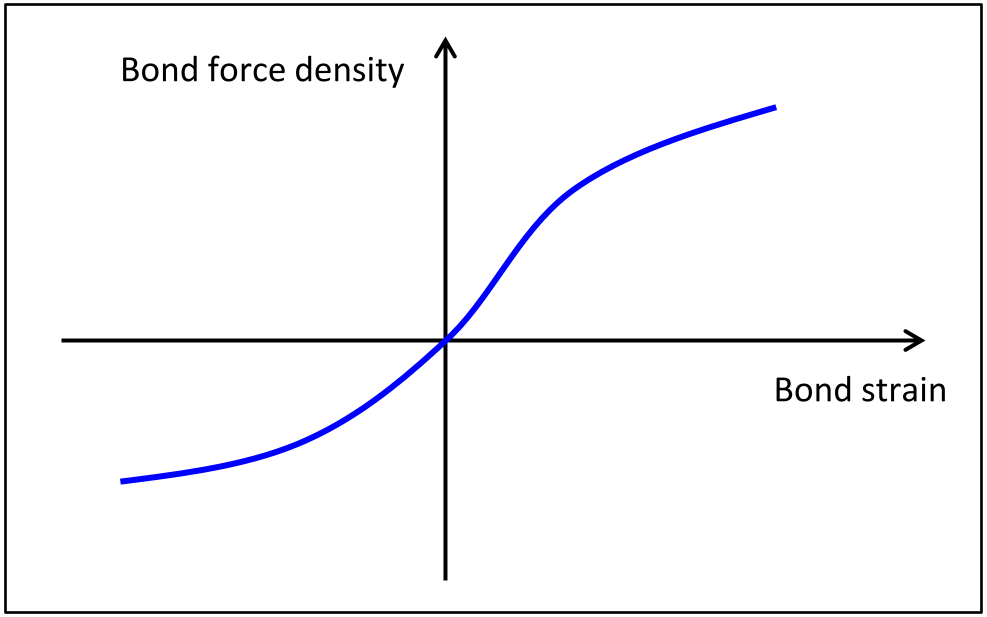
Peridynamics basics: What determines bond forces?

- Each pairwise bond force vector $\mathbf{f}(\mathbf{q}, \mathbf{x}, t)$ is determined jointly by:
- the *collective* deformation of $\mathcal{H}_{\mathbf{x}}$, and
- the *collective* deformation of $\mathcal{H}_{\mathbf{q}}$.
- Bond forces are antisymmetric: $\mathbf{f}(\mathbf{x}, \mathbf{q}, t) = -\mathbf{f}(\mathbf{q}, \mathbf{x}, t)$.



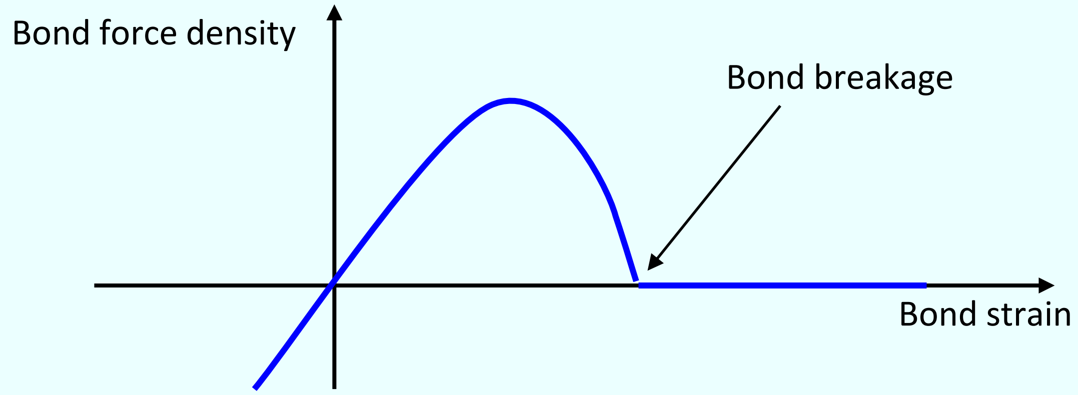
Bond based materials

- If each bond response is independent of the others, the resulting material model is called bond-based.
- The material model is then simply a graph of bond force density vs. bond strain.
- Main advantage: simplicity.
- Main disadvantage: restricts the material response.
    - Poisson ratio always = 1/4.



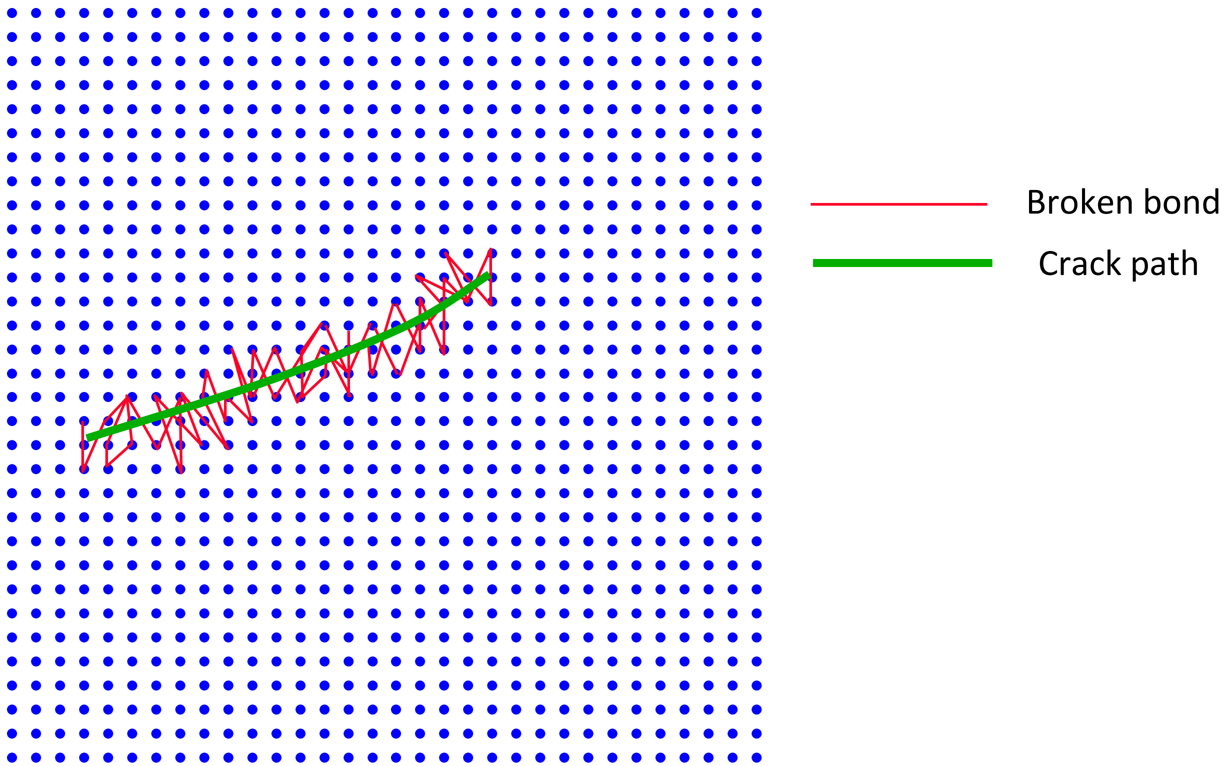
Damage due to bond breakage

- Recall: each bond carries a force.
- Damage is implemented at the bond level.
  - Bonds break irreversibly according to some criterion.
  - Broken bonds carry no force.
- Examples of criteria:
  - Critical bond strain (brittle).
  - Hashin failure criterion (composites).
  - Gurson (ductile metals).



Critical bond strain damage model

Autonomous crack growth



- When a bond breaks, its load is shifted to its neighbors, leading to progressive failure.

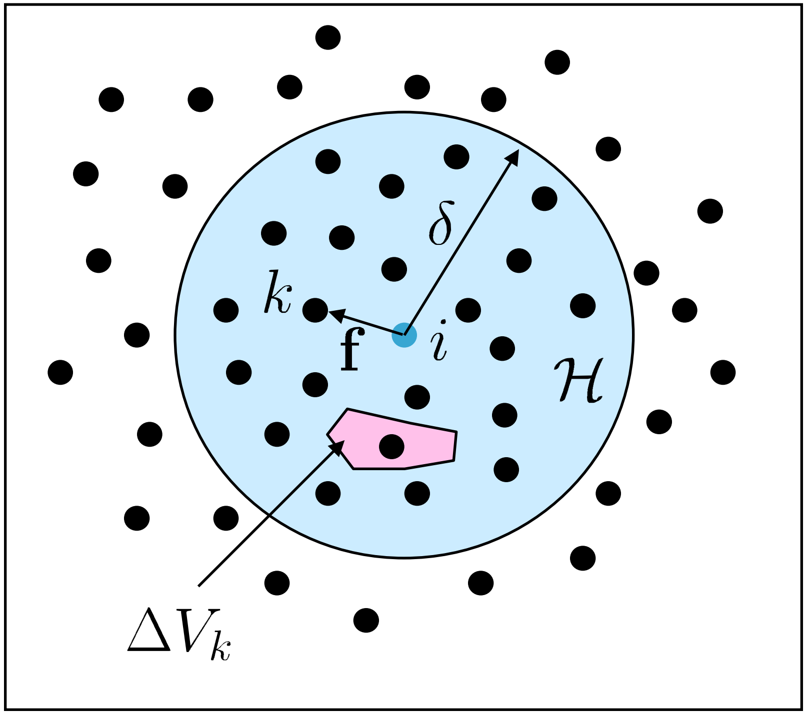
EMU numerical method

- Integral is replaced by a finite sum: resulting method is meshless and Lagrangian.

$$\rho\ddot{\mathbf{y}}(\mathbf{x},t) = \int_{\mathcal{H}} \mathbf{f}(\mathbf{x}',\mathbf{x},t)\, dV_{\mathbf{x}'} + \mathbf{b}(\mathbf{x},t)$$

$$\downarrow$$

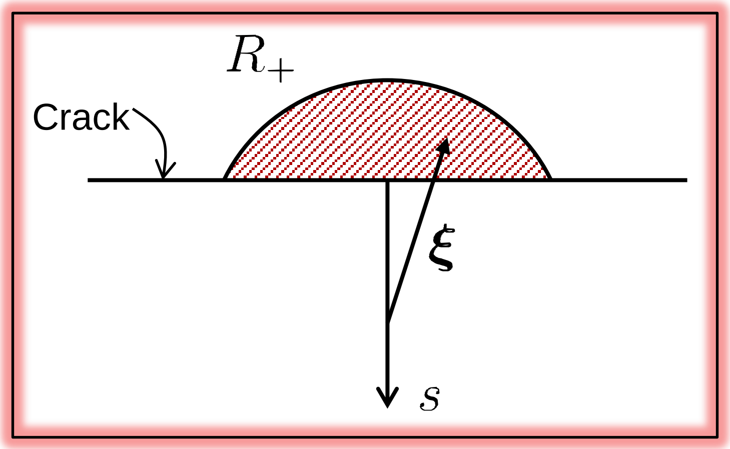
$$\rho\ddot{\mathbf{y}}_i^n = \sum_{k\in\mathcal{H}} \mathbf{f}(\mathbf{x}_k,\mathbf{x}_i,t)\, \Delta V_k + \mathbf{b}_i^n$$

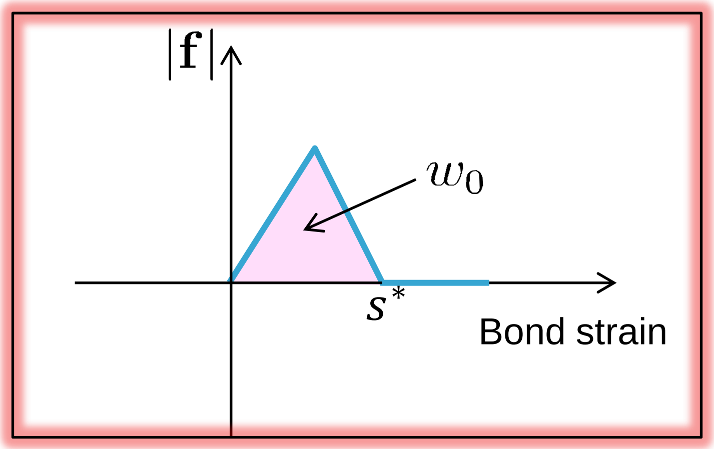


Critical bond strain: Relation to critical energy release rate

If the work required to break the bond $\boldsymbol{\xi}$ is $w_0(\boldsymbol{\xi})$, then the energy release rate is found by summing this work per unit crack area (J. Foster):

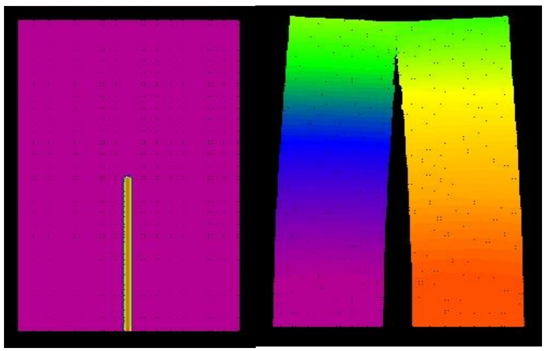
$$G = \int_0^\delta \int_{R_+} w_0(\boldsymbol{\xi})\, dV_{\boldsymbol{\xi}}\, ds$$

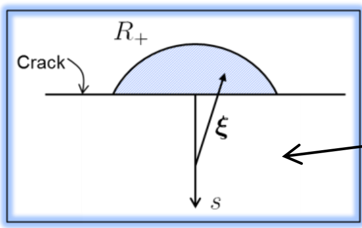




- Can then get the critical strain for bond breakage s^* in terms of G.
- Could also use the peridynamic J-integral as a bond breakage criterion.

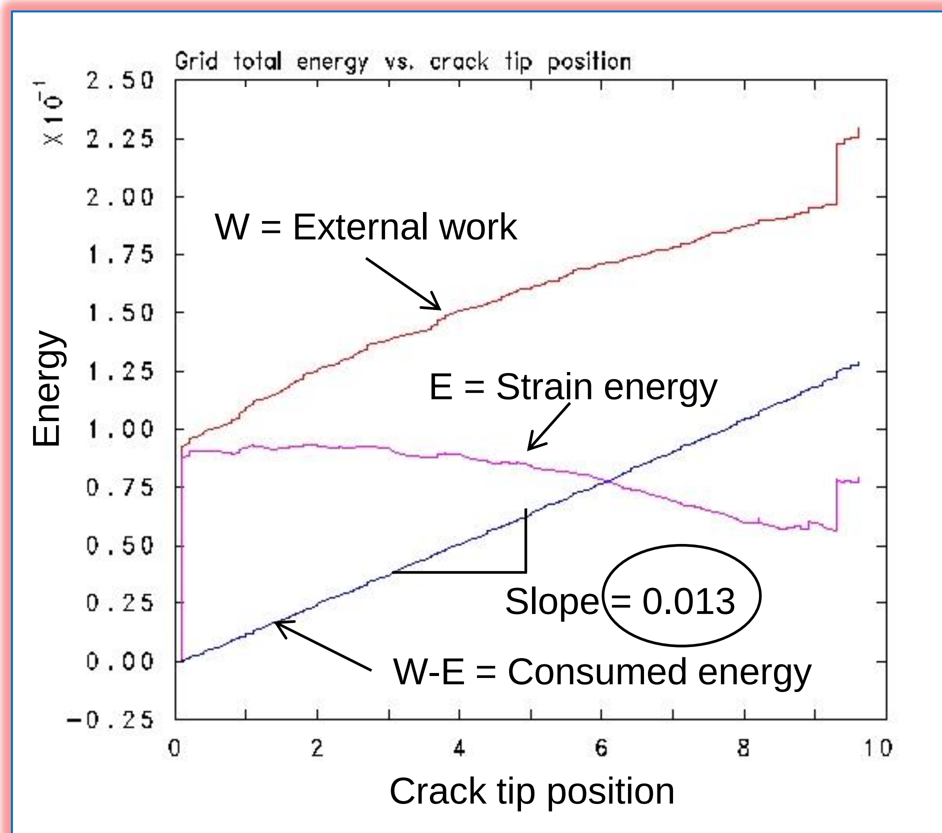
Energy balance for a crack: validation





From bond properties, energy release rate should be

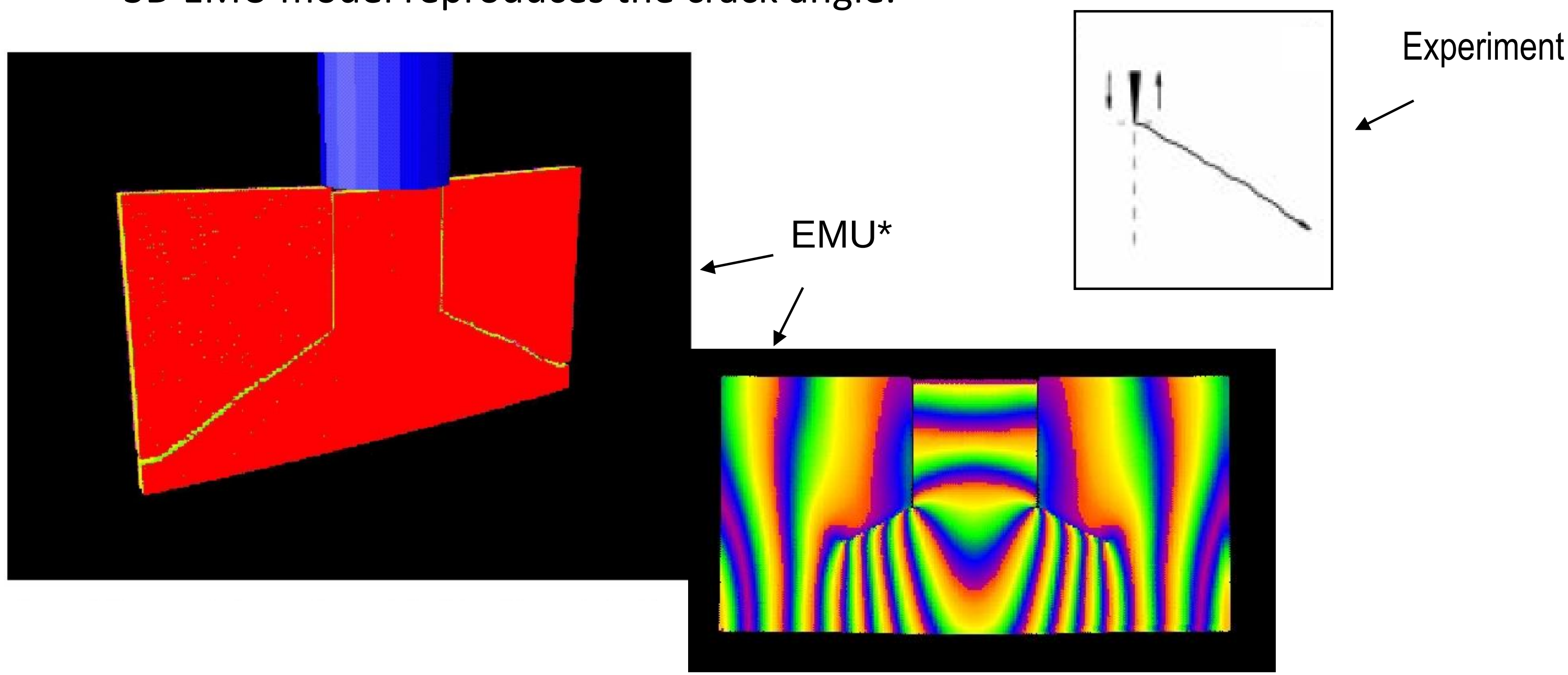
G = 0.013



- This confirms that the energy consumed per unit crack growth area equals the expected value from bond breakage properties.

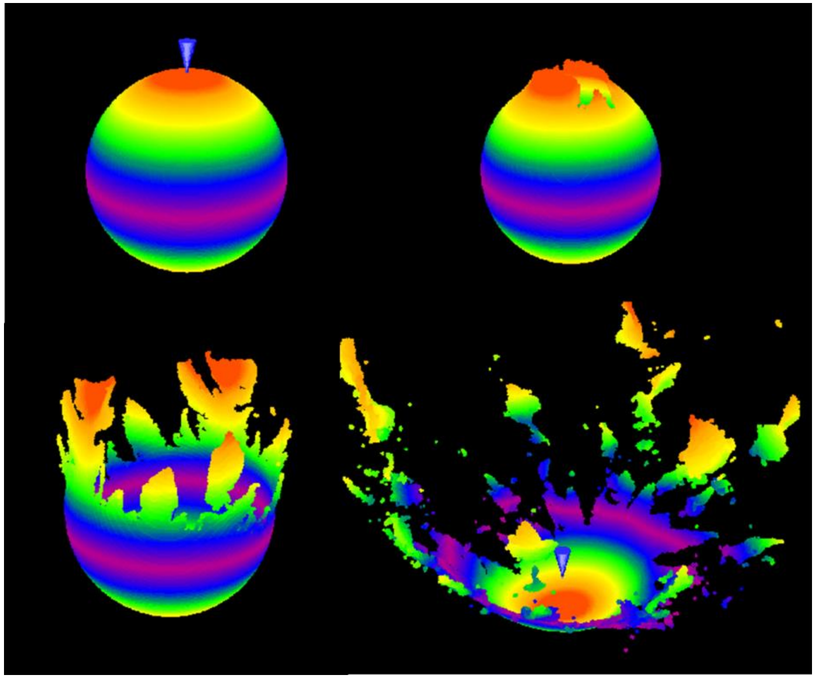
Dynamic fracture in a hard steel plate

- Dynamic fracture in maraging steel (Kalthoff & Winkler, 1988)
    - Mode-II loading at notch tips results in mode-I cracks at 70deg angle.
    - 3D EMU model reproduces the crack angle.

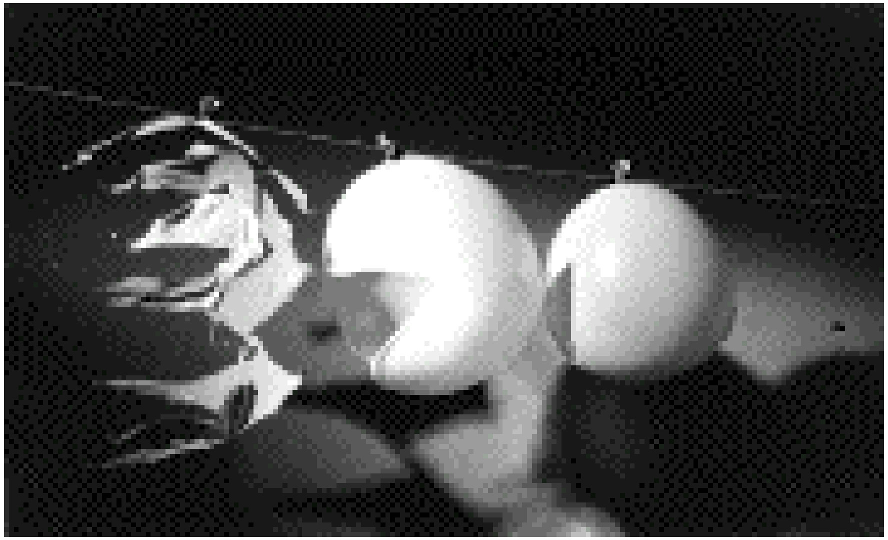


S. A. Silling, Dynamic fracture modeling with a meshfree peridynamic code, in *Computational Fluid and Solid Mechanics 2003*, K.J. Bathe, ed., Elsevier, pp. 641–644.

Dynamic fracture in membranes



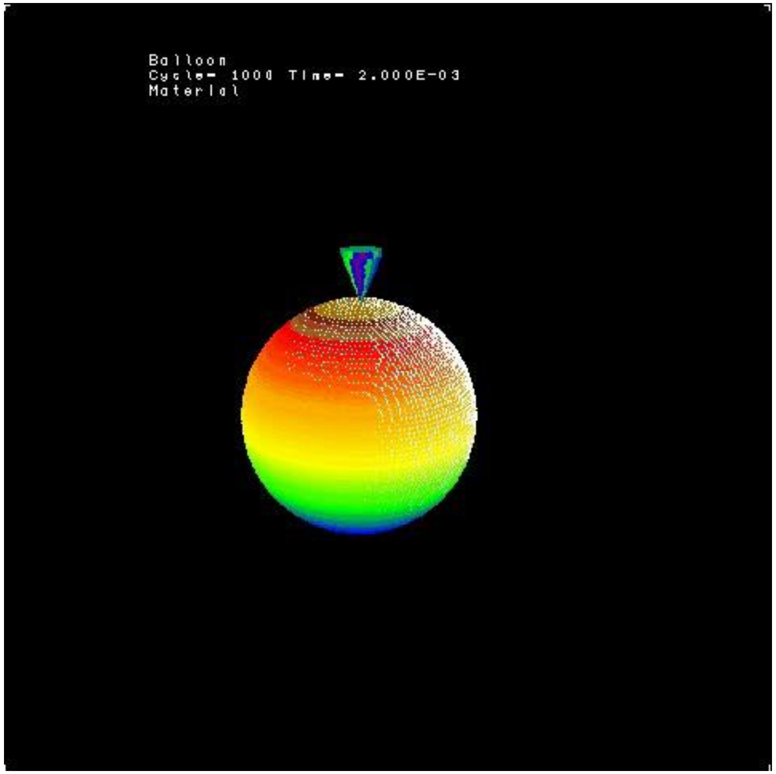
EMU model of a balloon penetrated
by a fragment



Early high speed photograph by Harold Edgerton
(MIT collection)
http://mit.edu/6.933/www/Fall2000/edgerton/edgerton.ppt

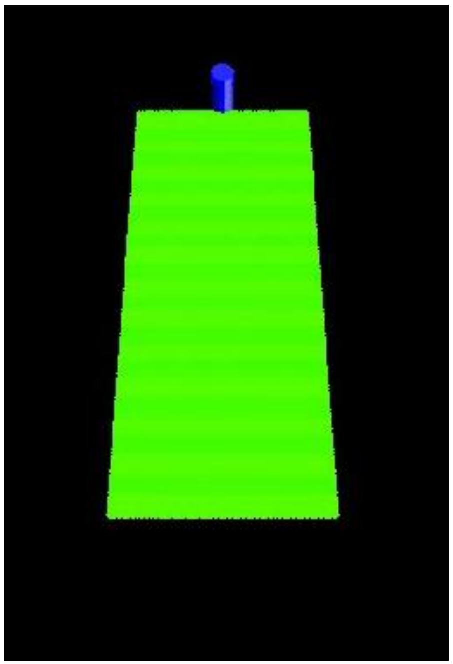
Pressurized shell struck by a fragment

Video

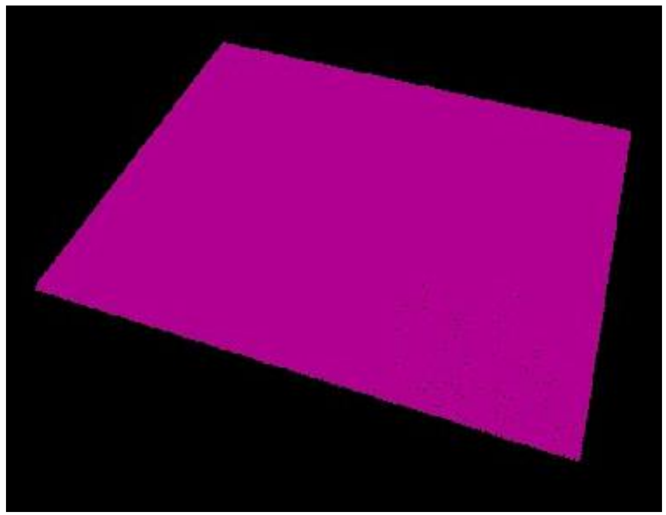


Examples: Membranes and thin films

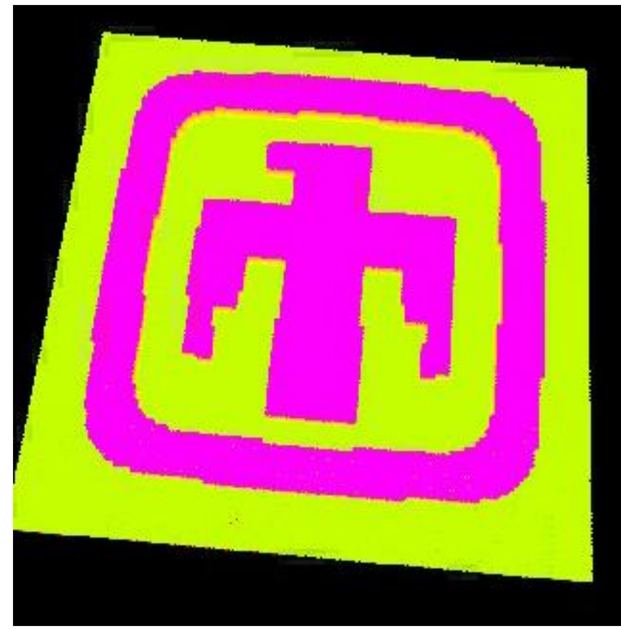
Videos



Oscillatory crack path

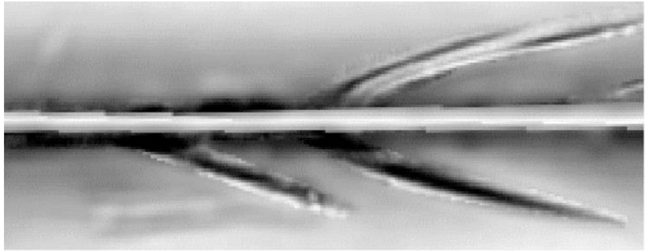


Crack interaction in a sheet

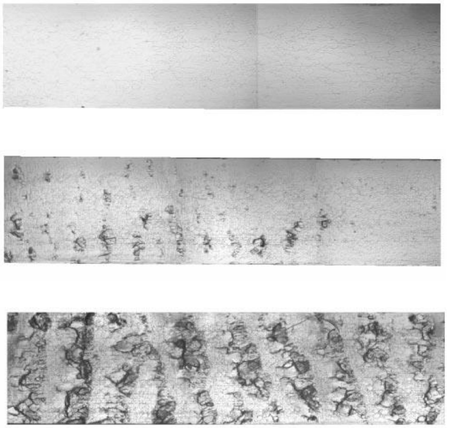


Aging of a film

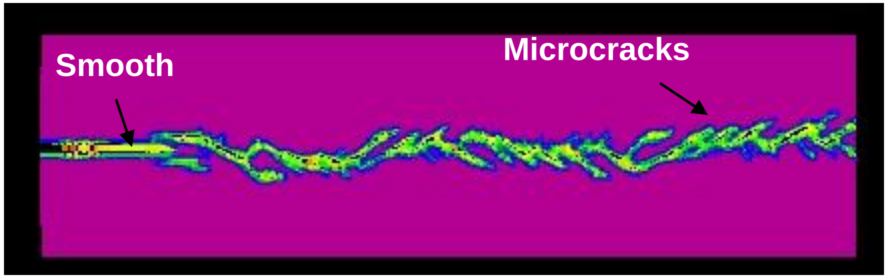
Dynamic fracture in PMMA: Damage features



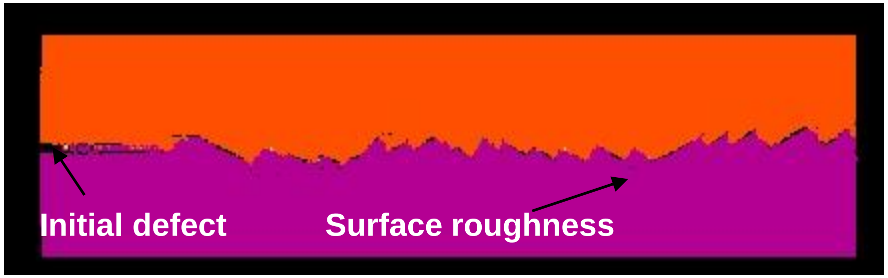
Microbranching



Mirror-mist-hackle transition*



EMU damage

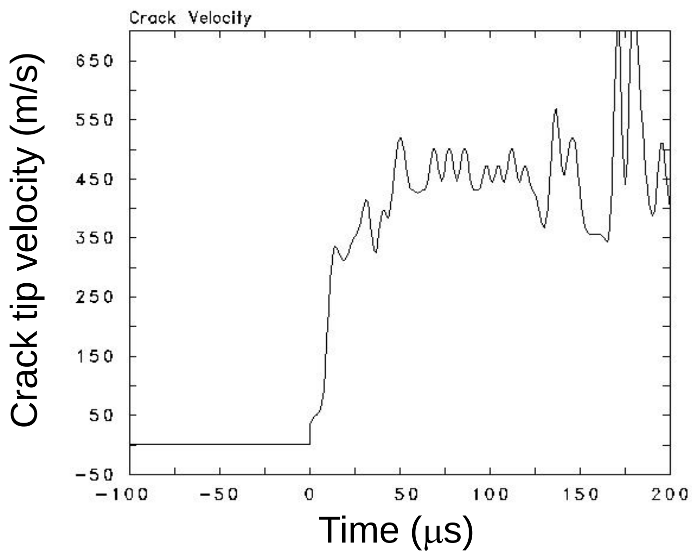


EMU crack surfaces

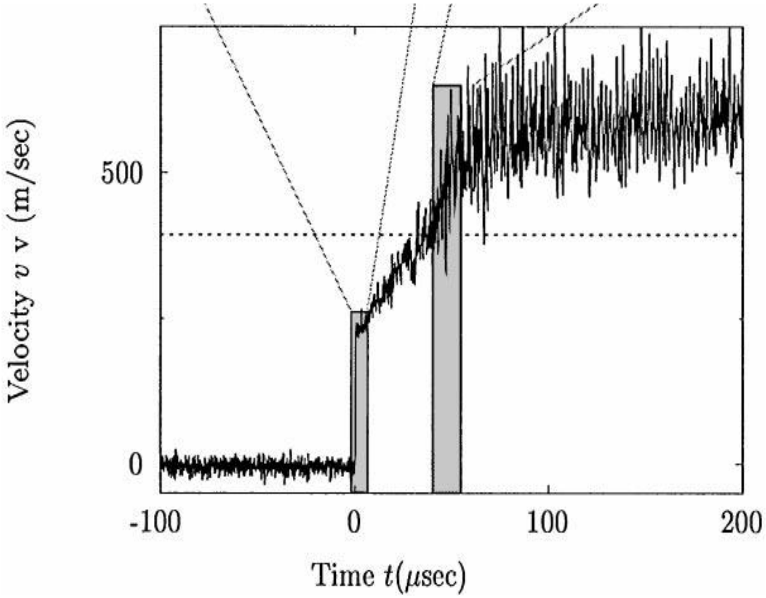
* J. Fineberg & M. Marder, *Physics Reports* 313 (1999) 1-108

Dynamic fracture in PMMA: Crack tip velocity

- Crack velocity increases to a critical value, then oscillates.



EMU

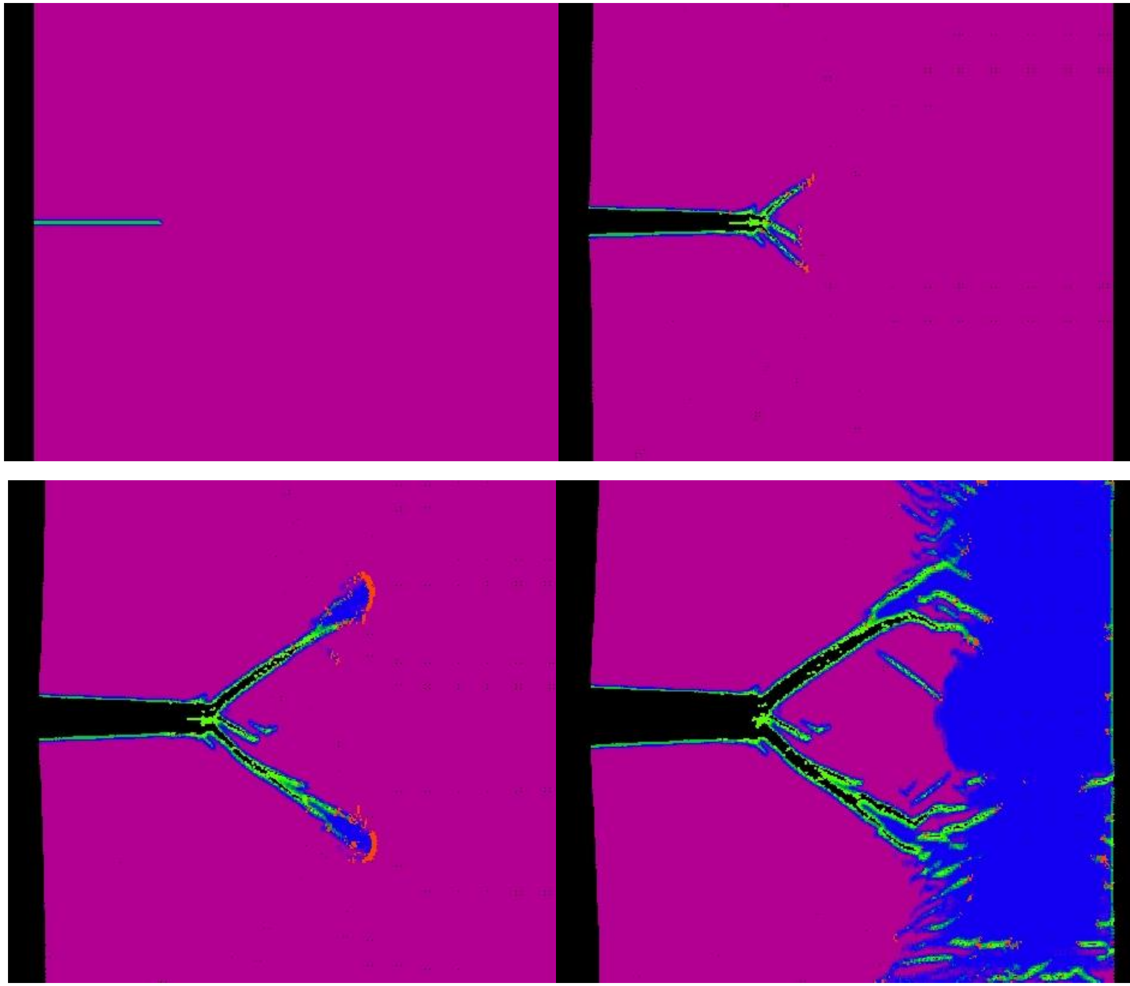


Experiment*

* J. Fineberg & M. Marder, *Physics Reports* 313 (1999) 1-108

Dynamic crack branching

- Similar to previous example but with higher strain rate applied at the boundaries.
- Red indicates bonds currently undergoing damage.
    - These appear ahead of the visible discontinuities.
- Blue/green indicate damage (broken bonds).
- More and more energy is being built up ahead of the crack – it can't keep up.
    - Leads to fragmentation.



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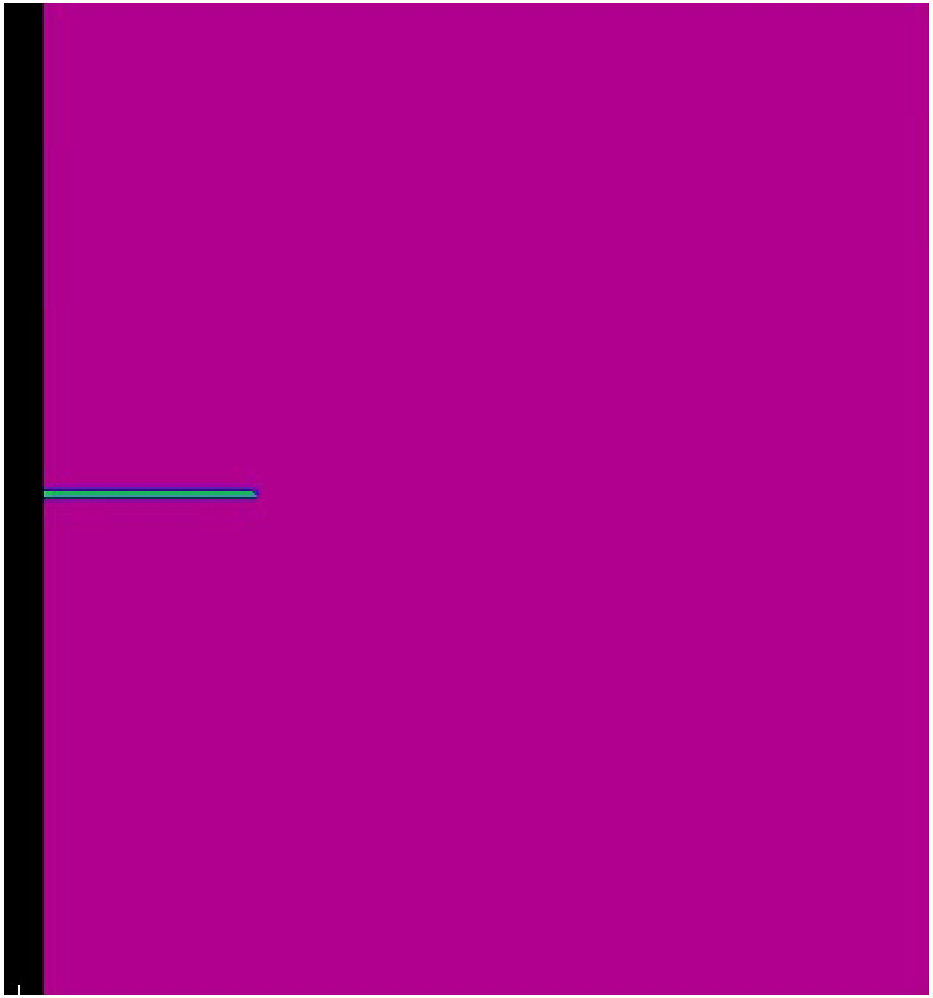
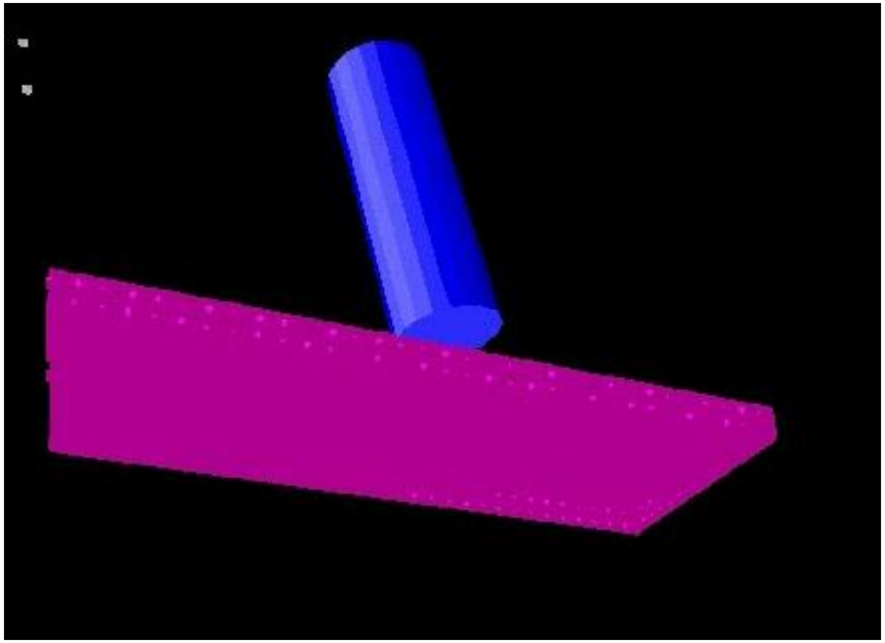


Video

Example:
Impact on reinforced concrete

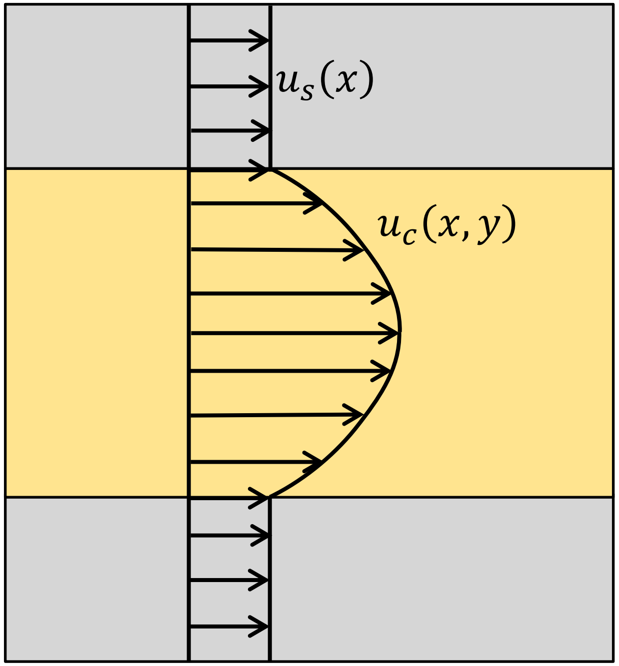
Video



Nonlocality – is it real?

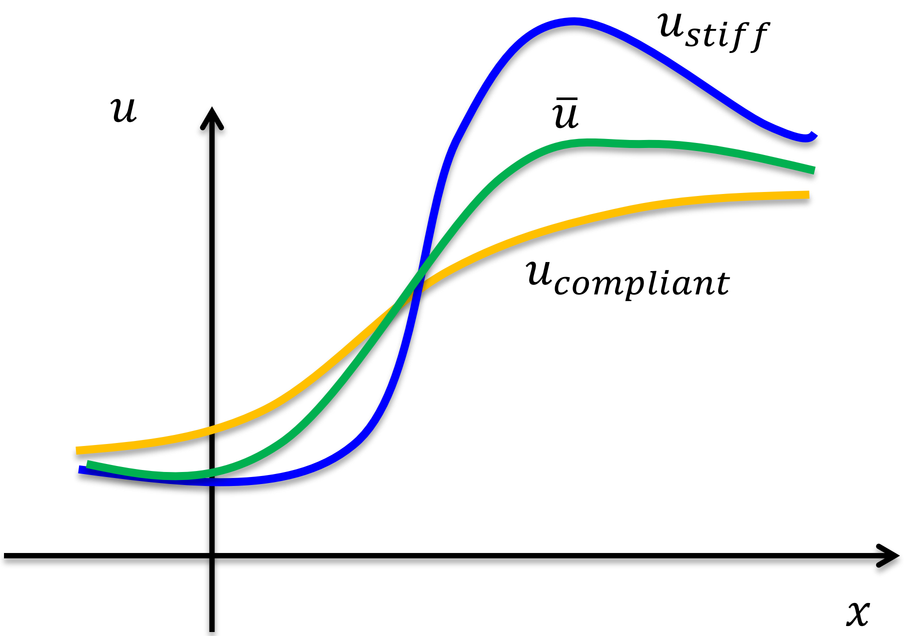
- It is commonly assumed that the local model (PDE-based) is an excellent approximation for continuous media, due to the small size of interatomic distances.
- This is true if we model the system in sufficient detail.
- When we use a "smoothed out" displacement field, nonlocality appears in the equations. Example…

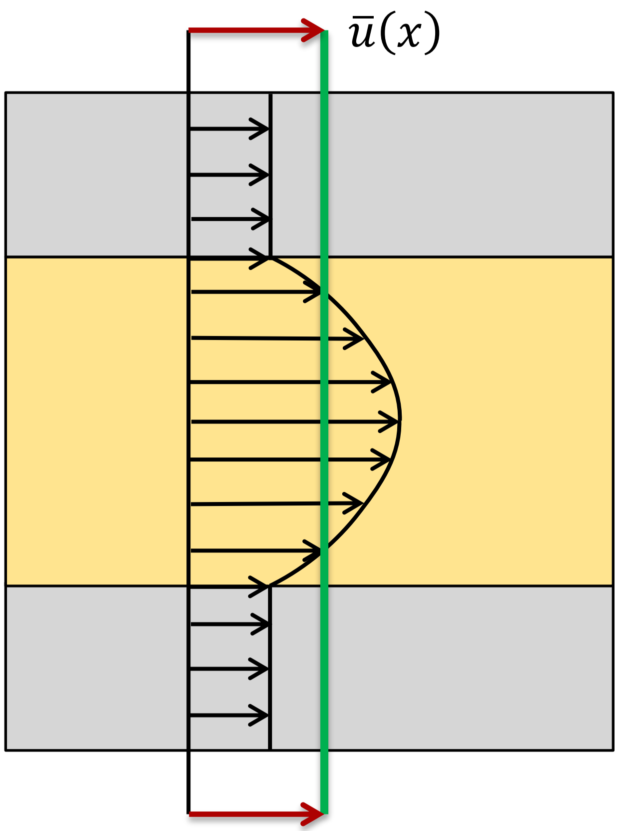
Layered composite (1D)



Nonlocality in a homogenized model

- Choose to model the composite as a single mass-weighted average displacement field $\bar{u}(x)$.





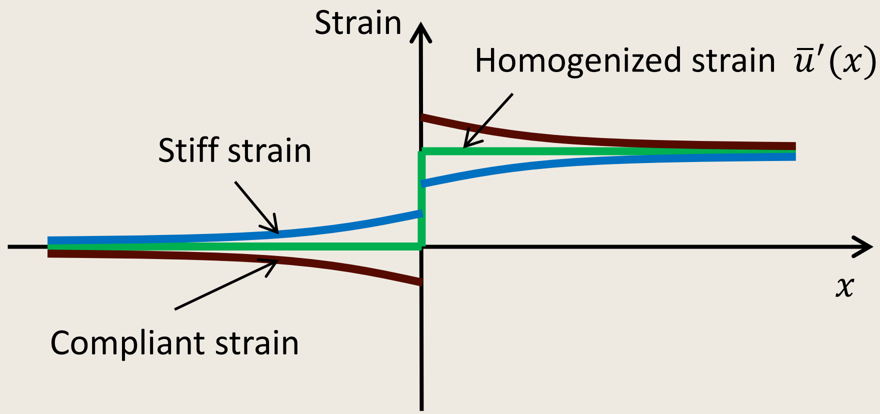
Nonlocality in a homogenized model

- After computing the force transfer between the phases, the equation of motion turns out to be

$$\rho\ddot{\bar{u}}(x,t) = E_c\bar{u}''(x,t) + \gamma k\lambda^4 \int_{-\infty}^{\infty} (\bar{u}(p,t) - \bar{u}(x,t))e^{-\lambda|x-p|}\, dp + b(x,t),$$

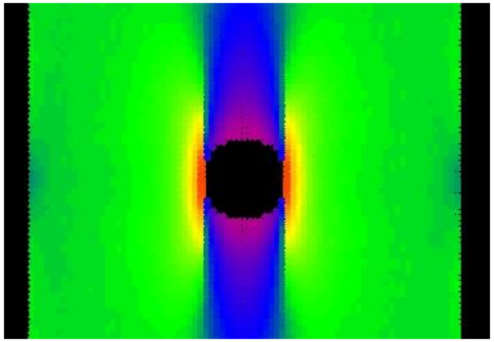
$$\frac{1}{\lambda} = \sqrt{\frac{E_s h_s h_c^2}{3\mu_c(h_s+h_c)}} = \text{length scale.}$$

Strain in each phase if the homogenized strain follows a step function

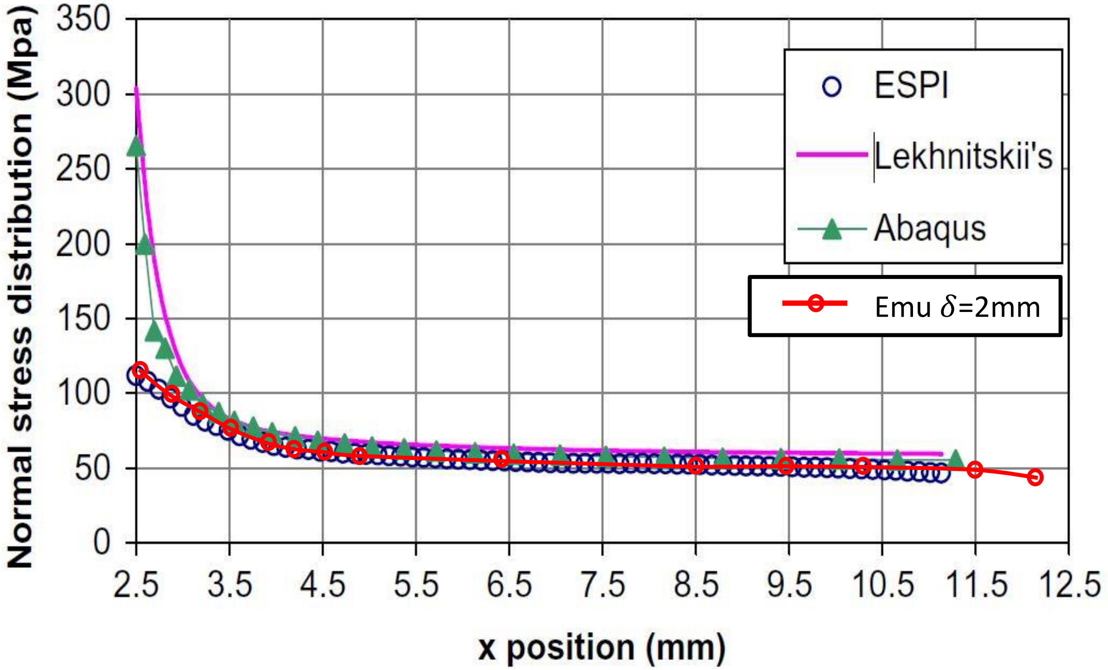


Are composites nonlocal?

- Peridynamic model is more accurate than the local model for predicting stress concentration in a laminate.
- $h_s = h_c$ =0.4mm, E_s = 150GPa, μ_c =4GPa.
- $\Rightarrow 1/\lambda = 1.41$mm.



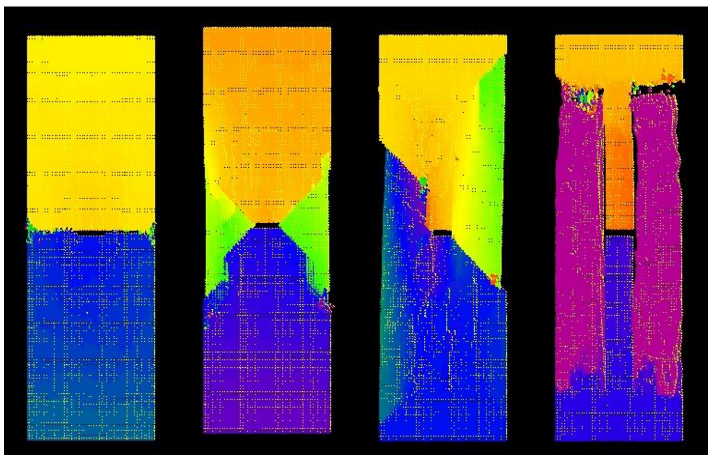
EMU: contours of longitudinal stress
Horizon = 2mm



Data of Toubal, Karama, and Lorrain, Composite Structures 68 (2005) 31-36

Splitting and fracture mode change in composites

- Distribution of fiber directions between plies strongly influences the way cracks grow.

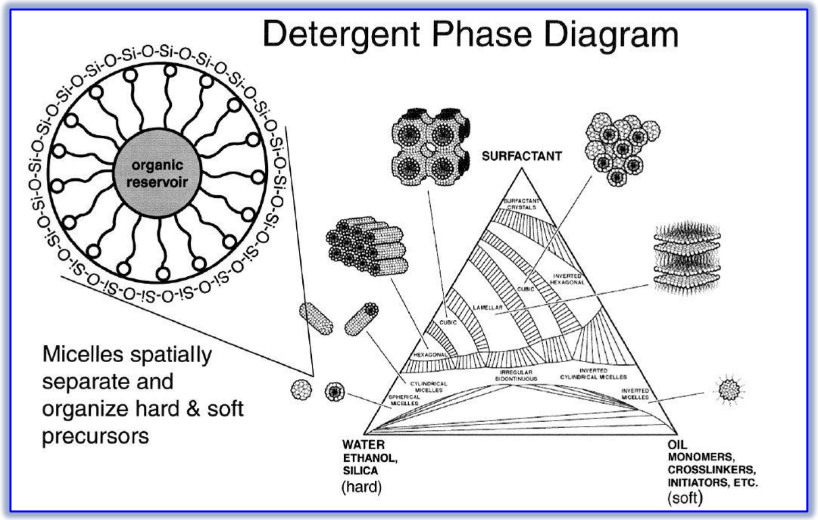


EMU simulations for different layups

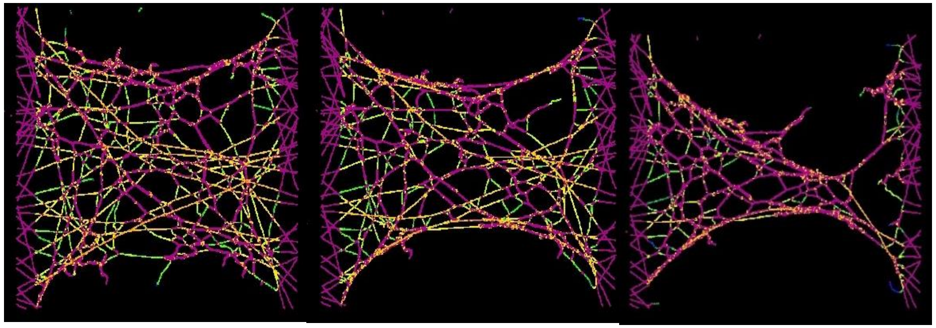
Typical crack growth in a notched laminate (photo courtesy Boeing)

Self-assembly and long-range forces

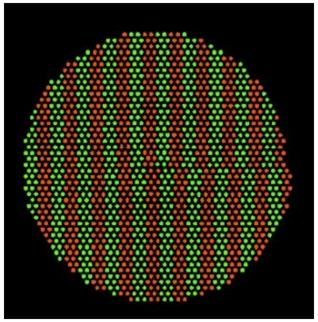
- Potential importance for self-assembled nanostructures.
- All forces are treated as long-range.



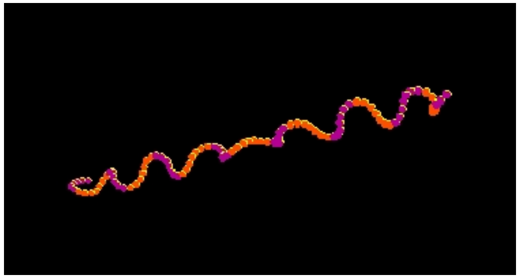
Self-assembly is driven by long-range forces
Image: Brinker, Lu, & Sellinger, Advanced Materials (1999)



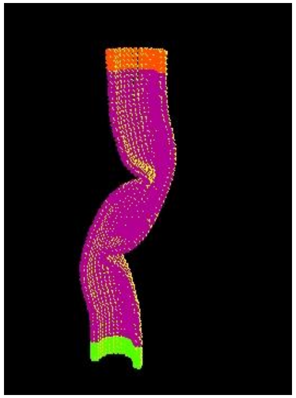
Failure in a nanofiber membrane
(F. Bobaru, Univ. of Nebraska)



Dislocation



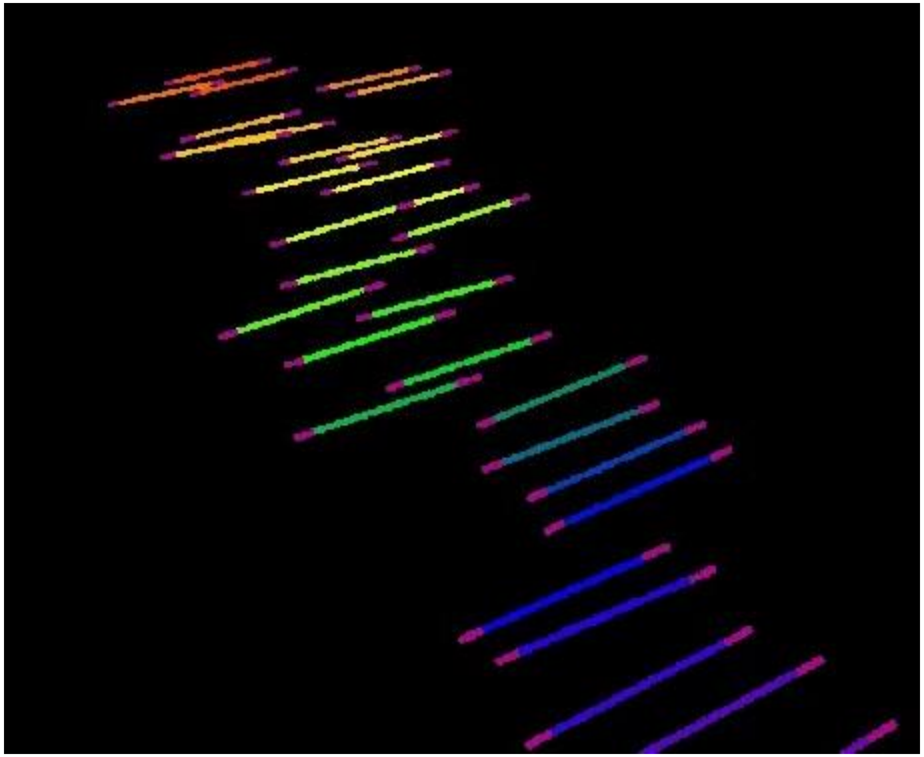
Nanofiber self-shaping



Carbon nanotube

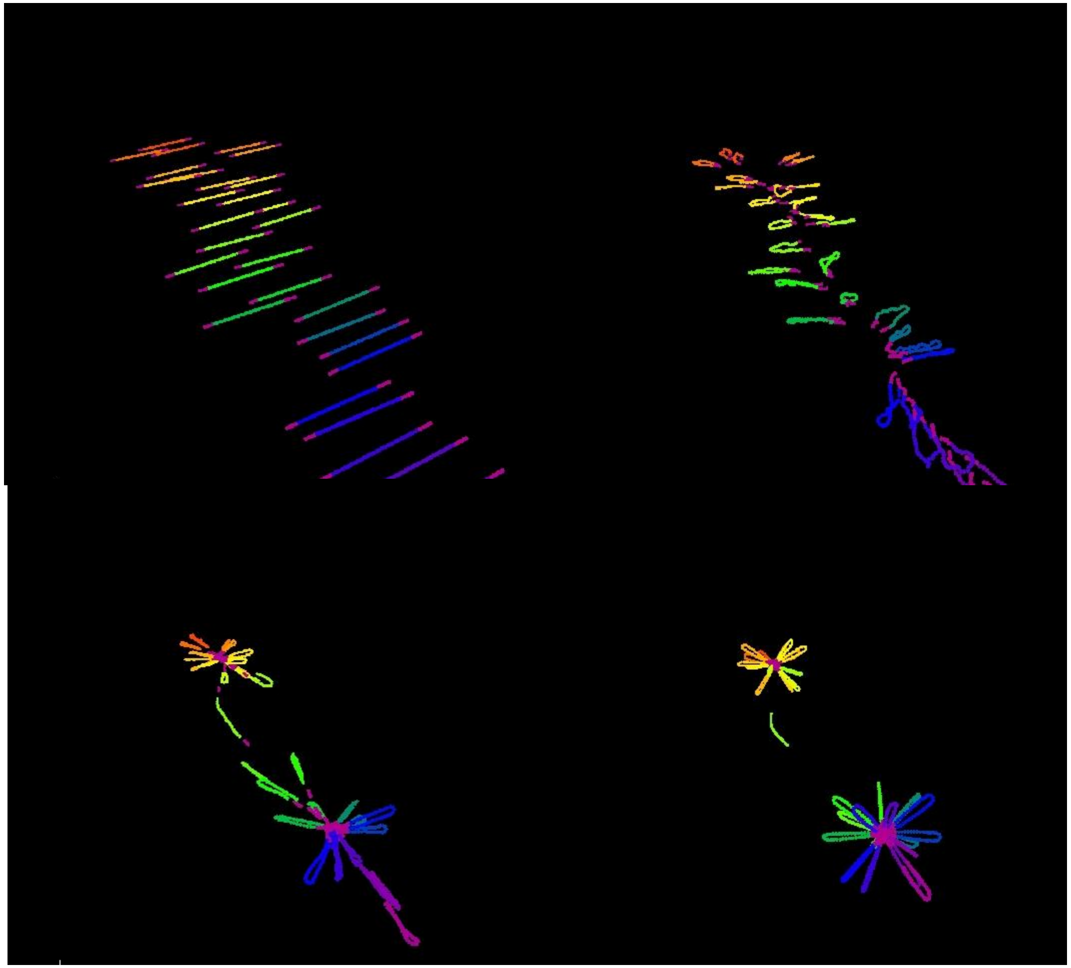
Self-assembly example

- Solution of long rods modeled as a peridynamic continuum:
  - Ends of the rods attract.
  - Inner parts of the rods repel.
  - Rods have a small resistance to bending.
- Rods are initially straight, then find a lower energy configuration.
- Peridynamics is useful because of the problem involves both continuum and long-range interactions.

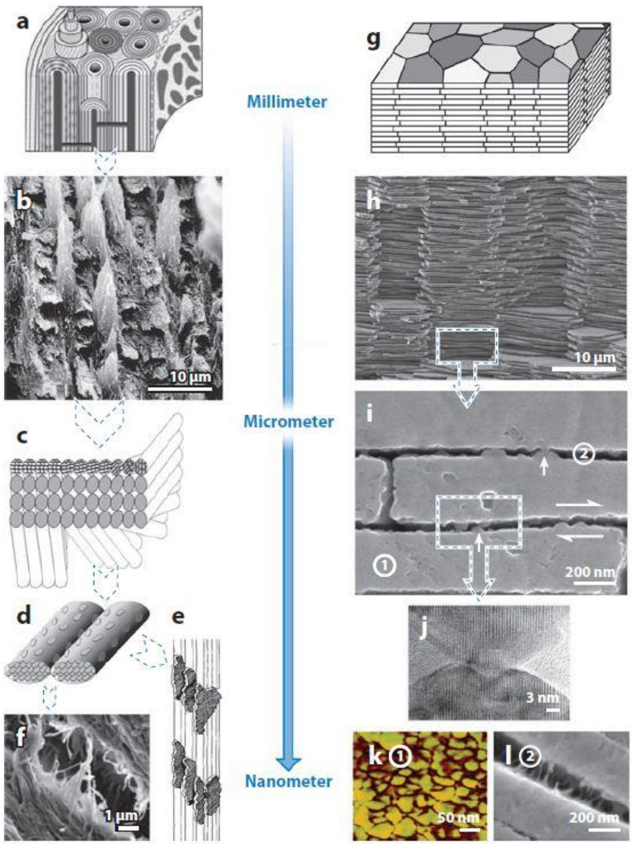


Video

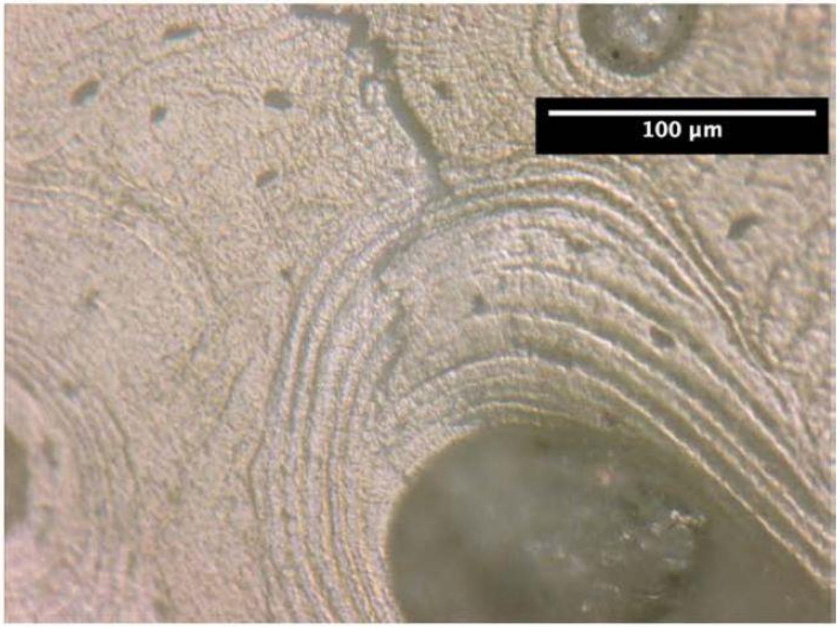
Self-assembly example



Bone: A composite material with many length scales



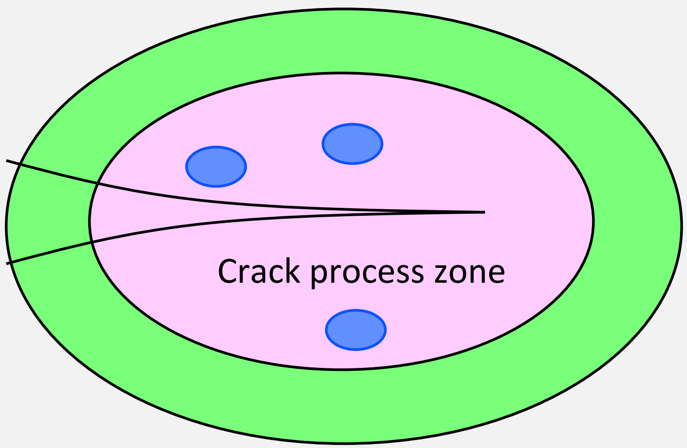
Bone contains a heirarchy of structures at many length scales. Image: Wang and Gupta, *Ann. Rev. Mat. Sci.* 41 (2011) 41-73



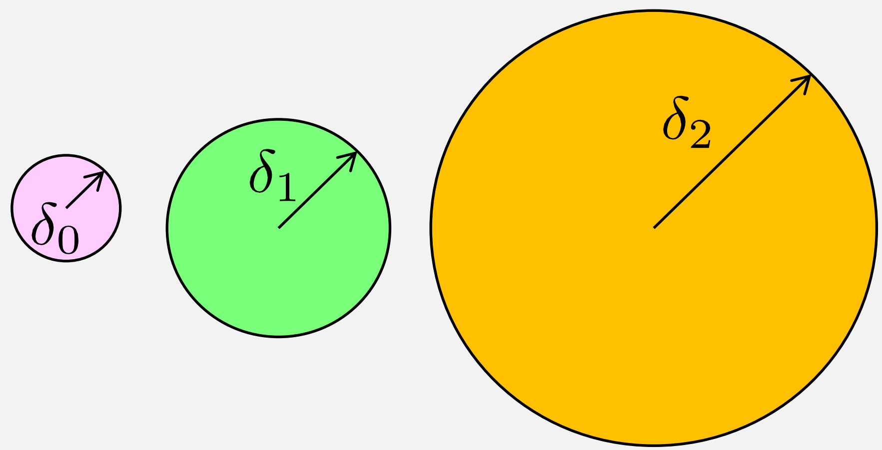
Bone structure helps delay, deflect crack growth. Image: Chan, Chan, and Nicolella, *Bone* 45 (2009) 427–434

Multiple length scales

- Objective: apply a suitable microscale model for processes near a crack tip at whatever length scale is dictated by physics.
- Method: hierarchy of models at different length scales.
  - Level 0: smallest.
  - Level > 0: coarsened.



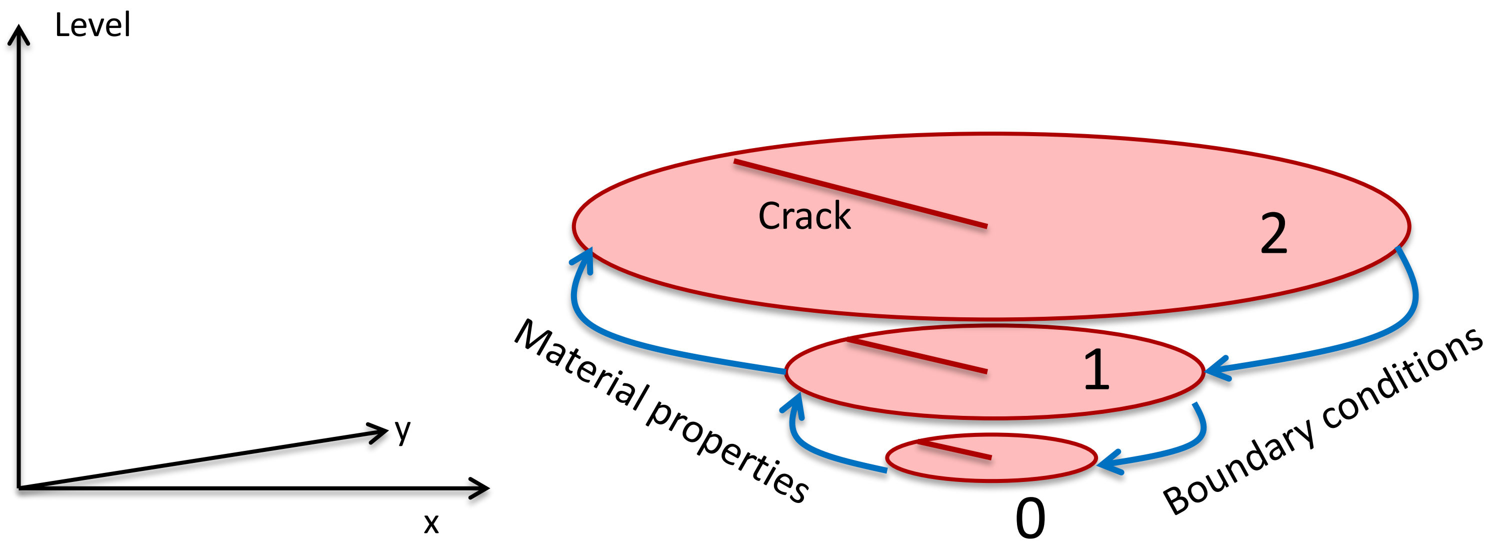
The details of damage evolution are always modeled at level 0.



Each successive level has a larger length scale (horizon).

Concurrent solution strategy

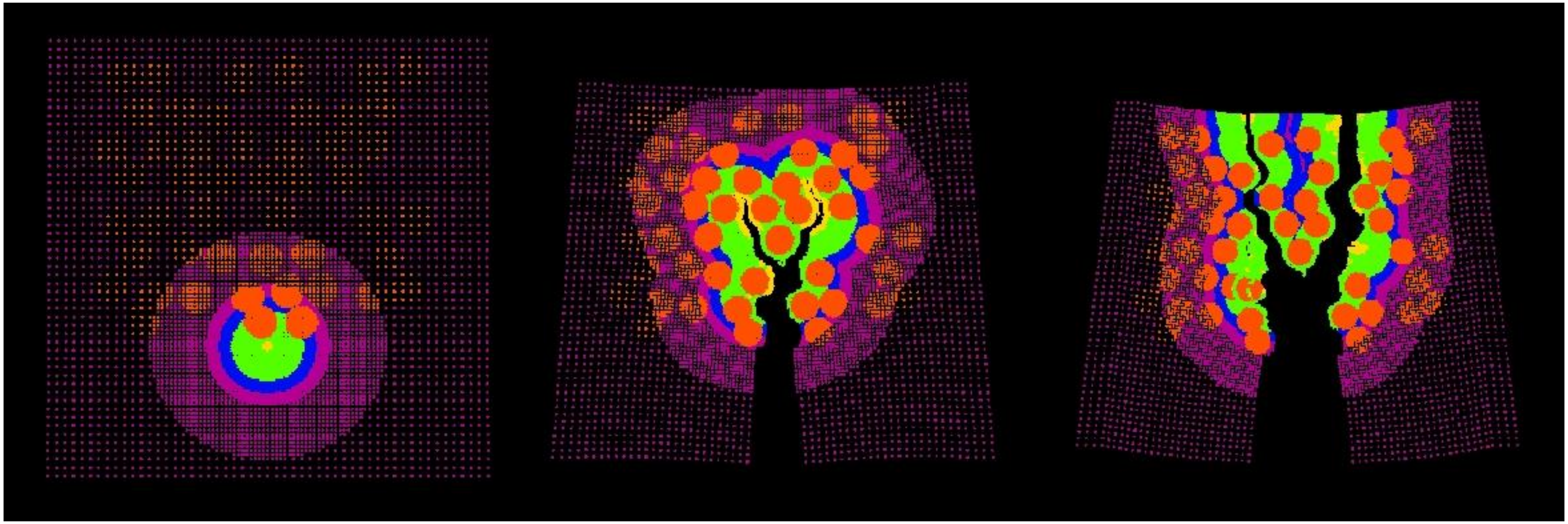
- The equation of motion is applied only <u>within</u> each level.
- Higher levels provide boundary conditions (really volume constraints) on lower levels.
- Lower levels provide coarsened material properties (including damage) to higher levels.



Schematic of communication between levels in a 2D body

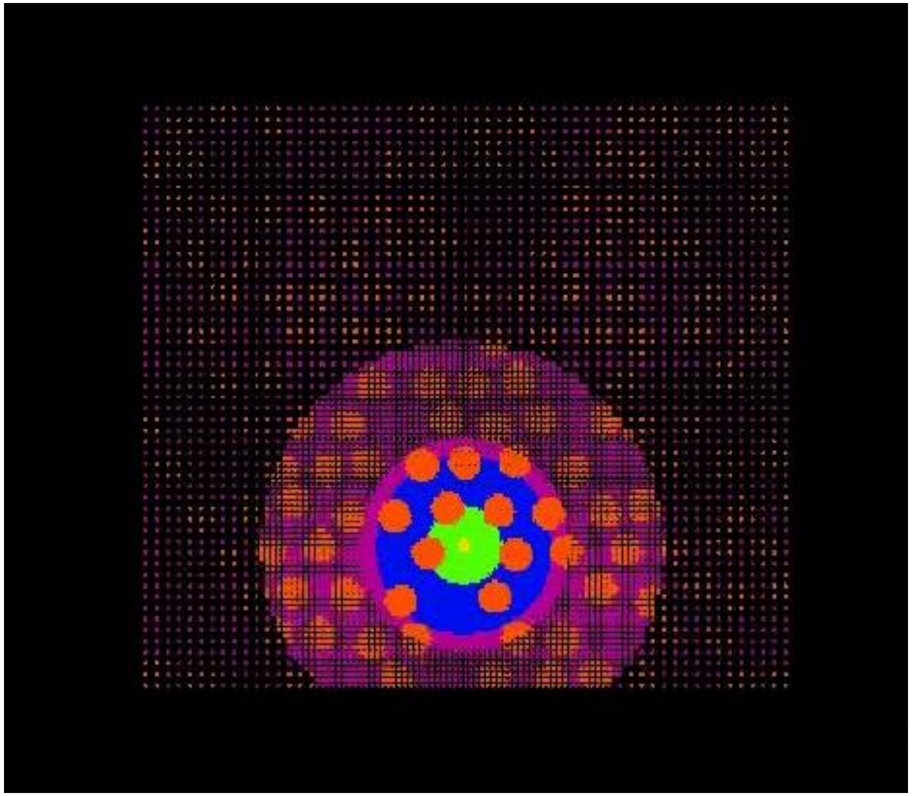
Branching in a heterogeneous medium

- Crack grows between randomly placed hard inclusions.



Heterogeneous medium

Video

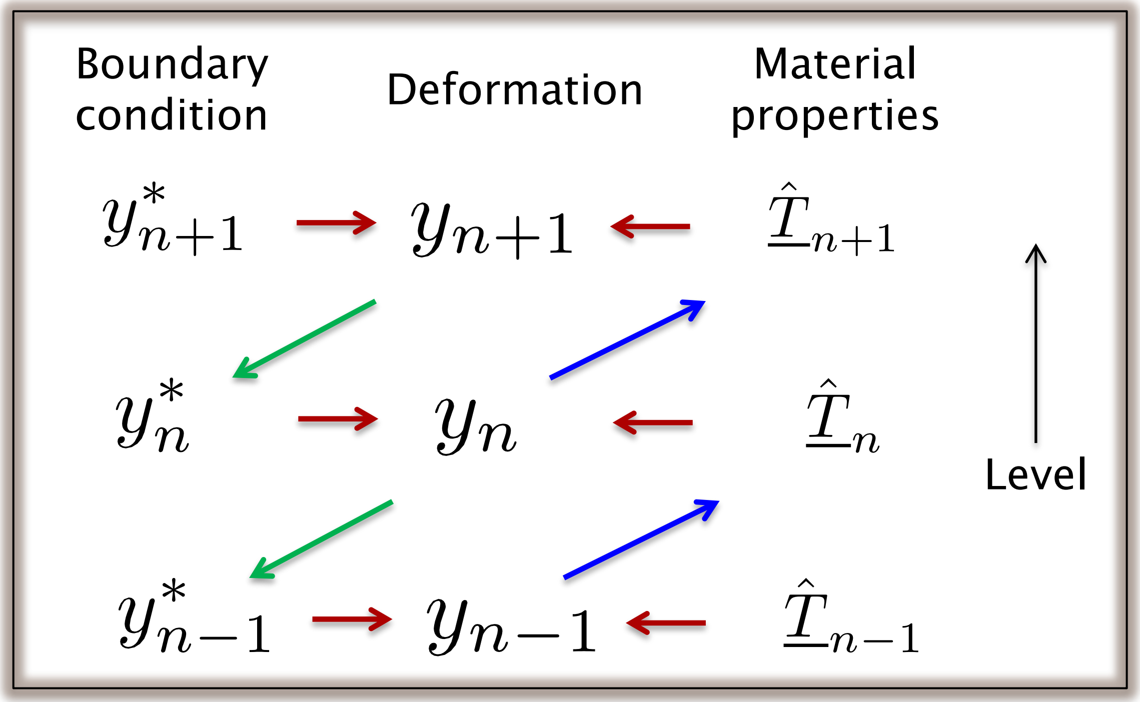


Discussion

- All forces are treated as long-range forces.
- The basic equations allow discontinuities – compatible with cracks.
- Cracks do whatever they want – no need for supplemental equations.
- Some practical difficulties:
  - Slower than standard finite elements.
  - Boundary conditions are different than in the standard theory.

Extra slides

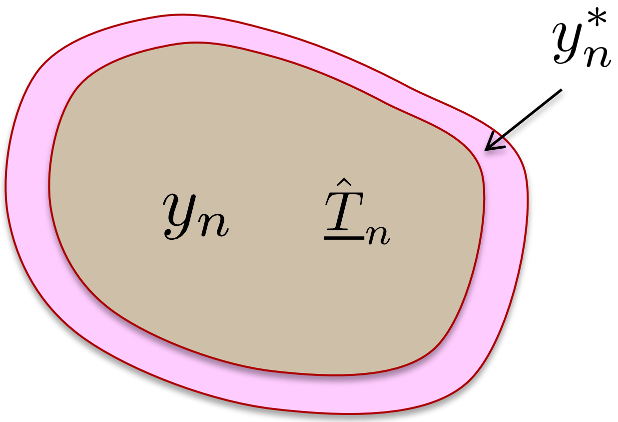
Dependencies between levels



Momentum balance

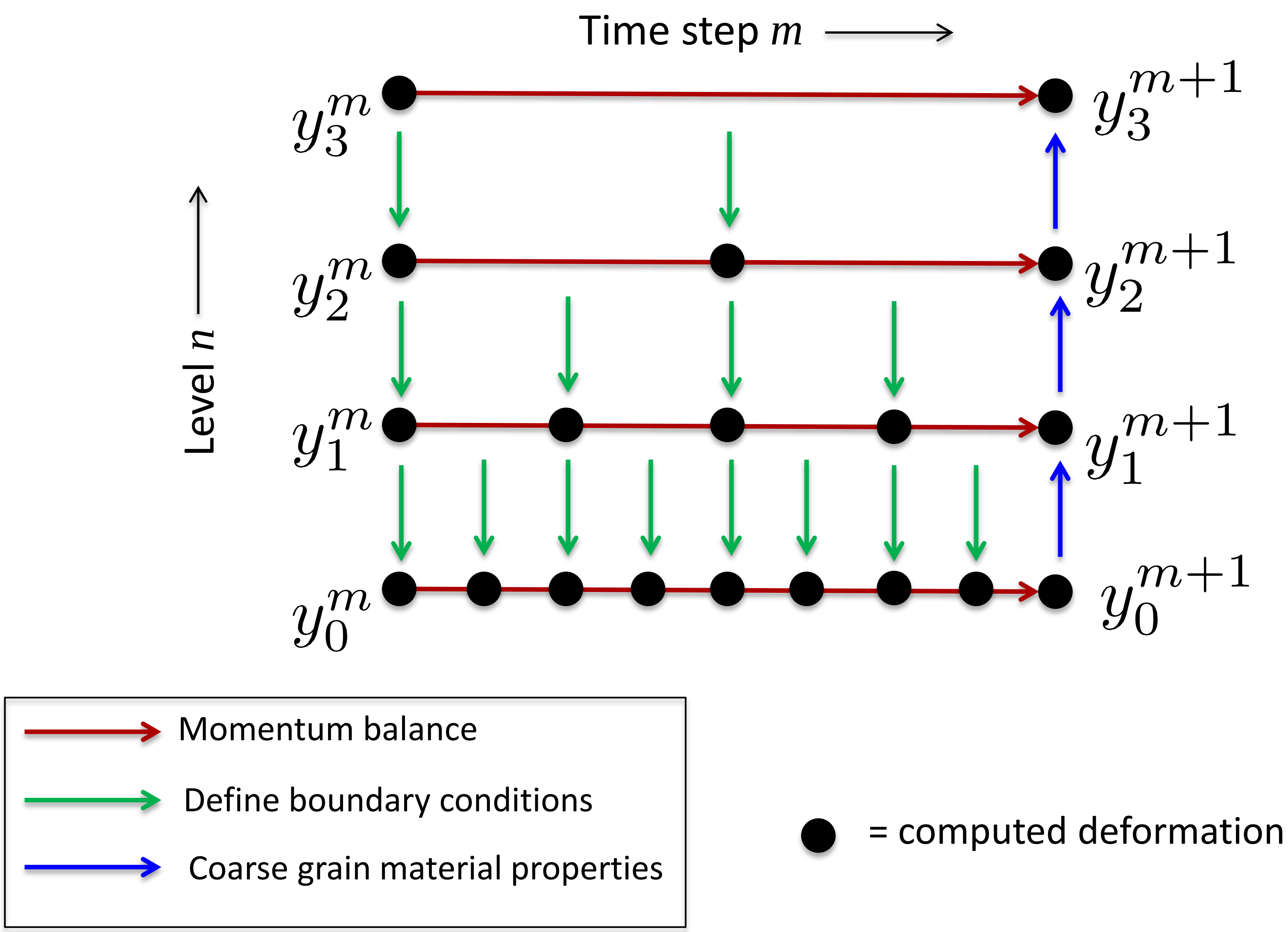
Define boundary conditions

Coarse grain material properties

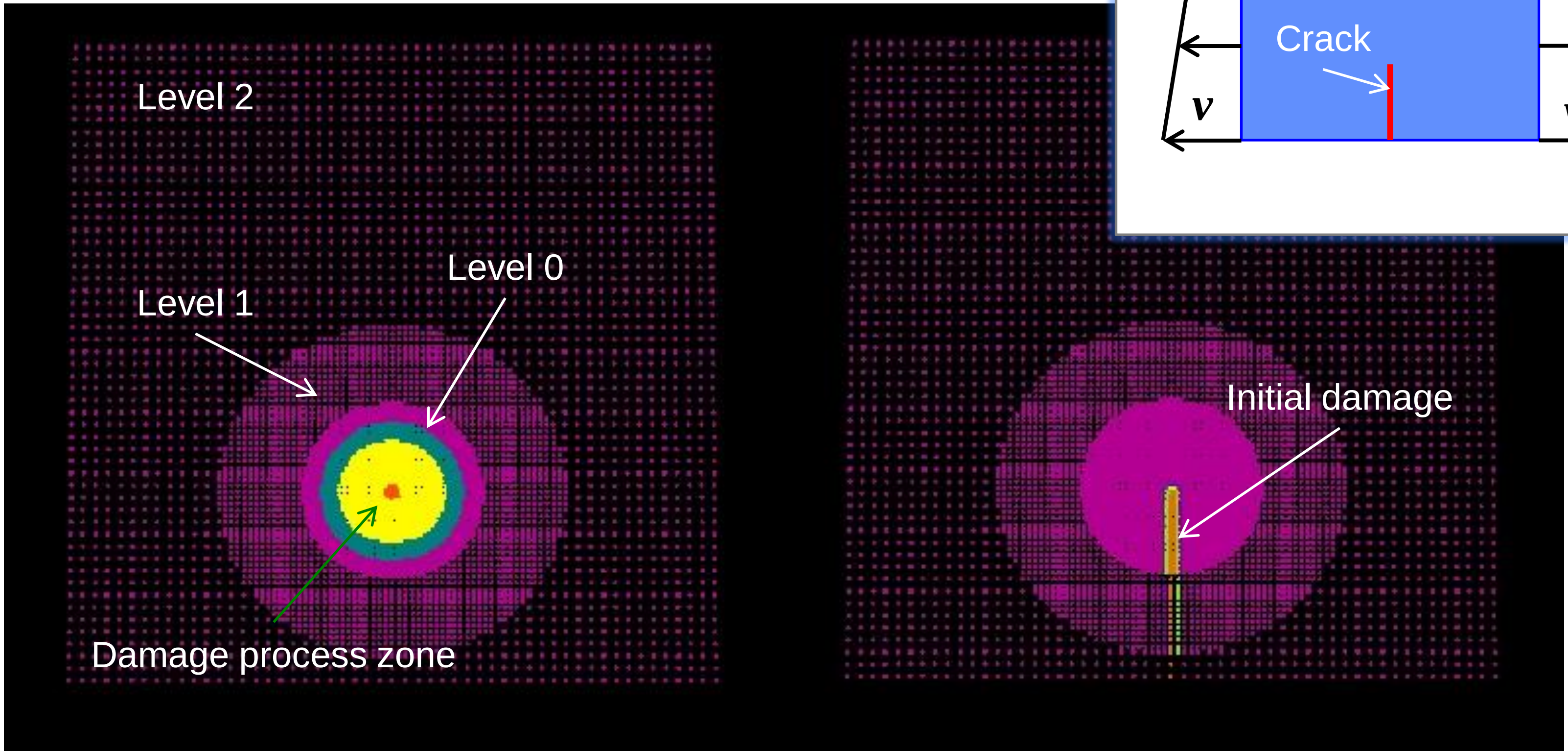


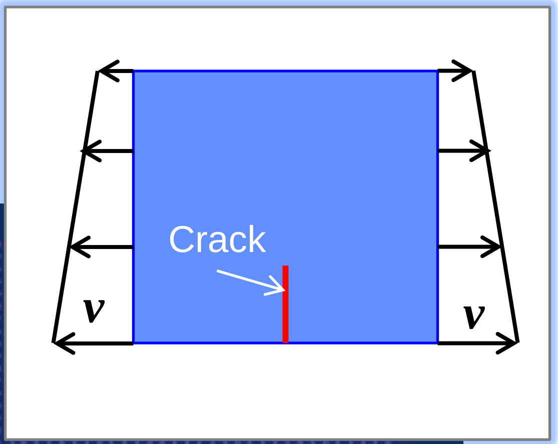
Level n problem

Flow of information in a time step

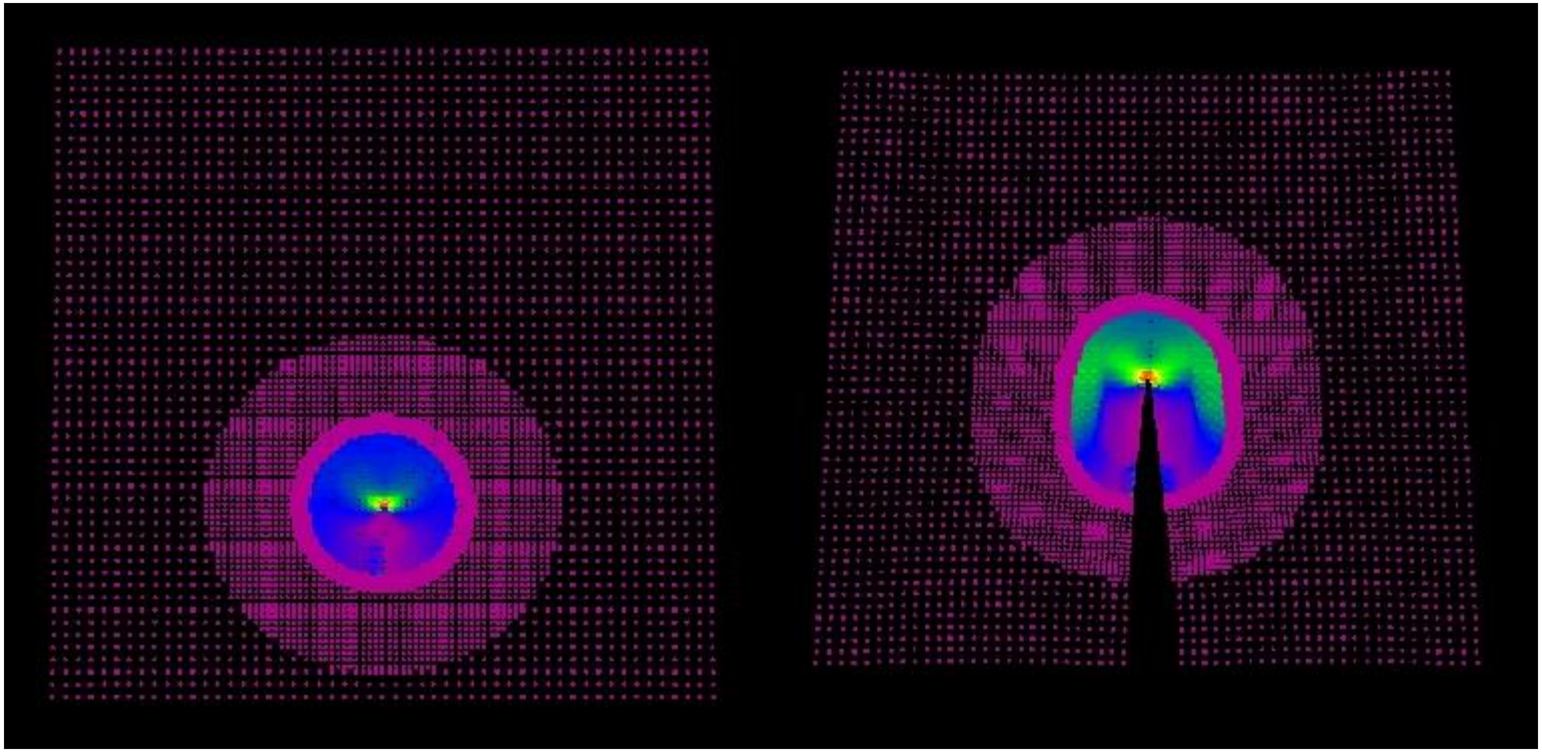


Multiscale examples: Crack growth in a brittle plate



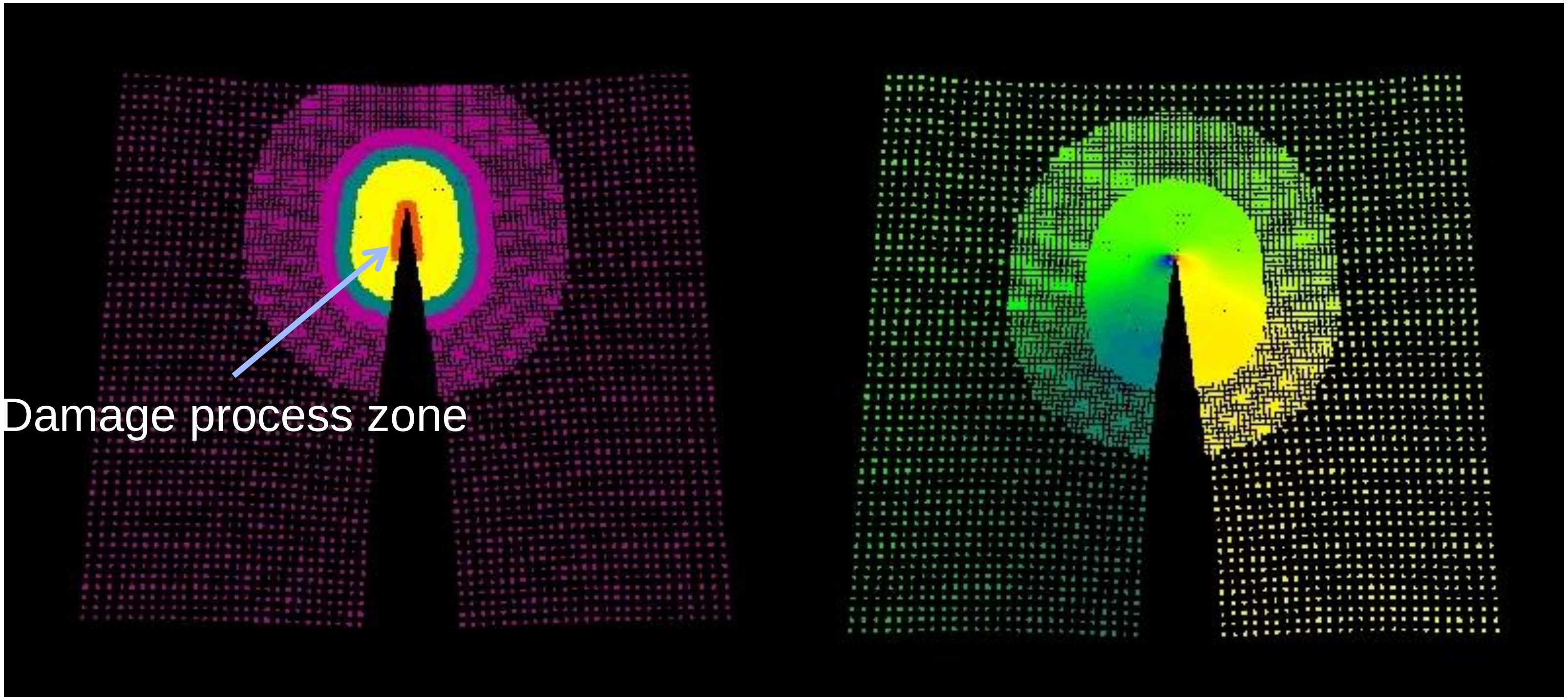


Crack growth in a brittle plate: Bond strains



Colors show the largest strain among all bonds connected to each node.

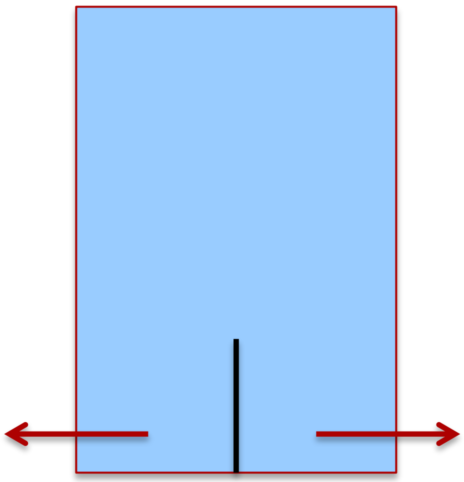
Levels move as the crack grows

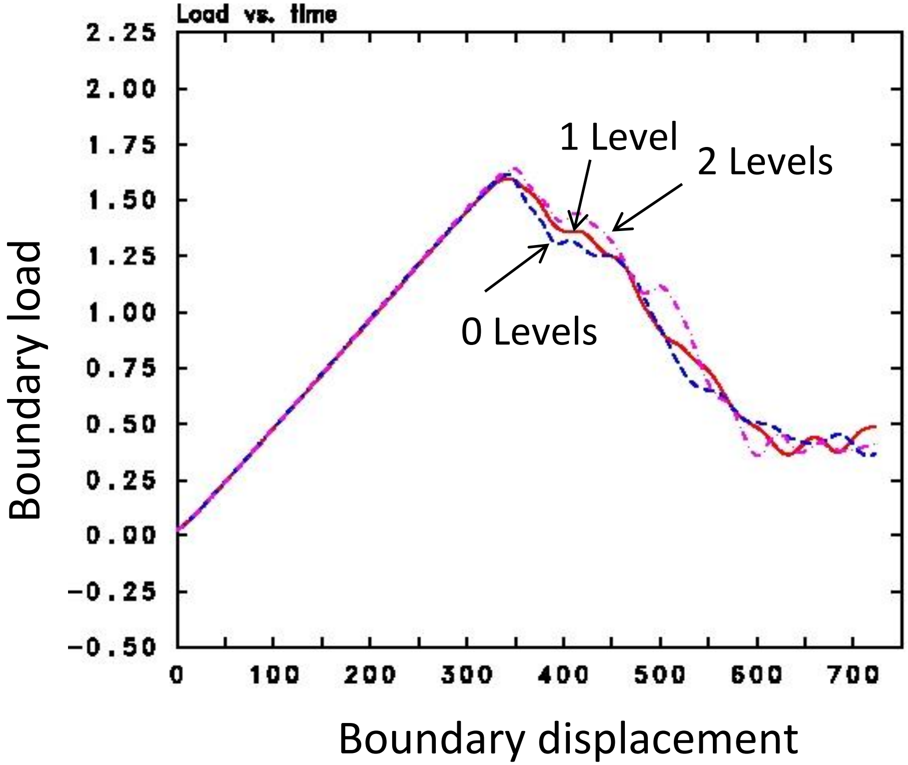


v_1 velocity

Results with and without multiscale

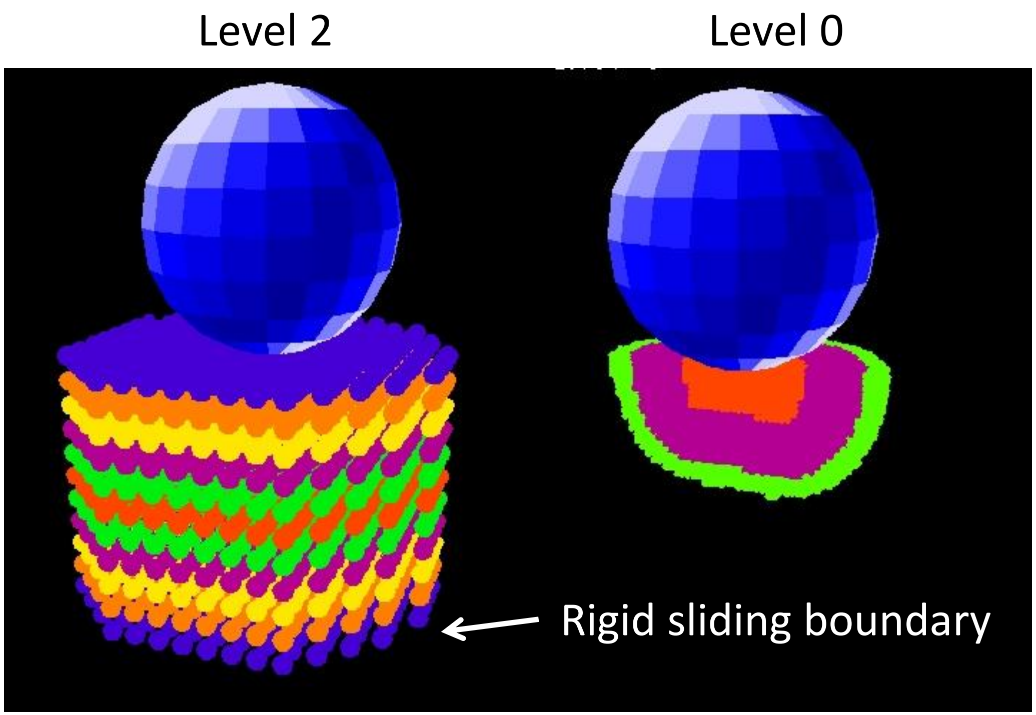
- All three levels give essentially the same answer.
- Higher levels substantially reduce the computational cost.



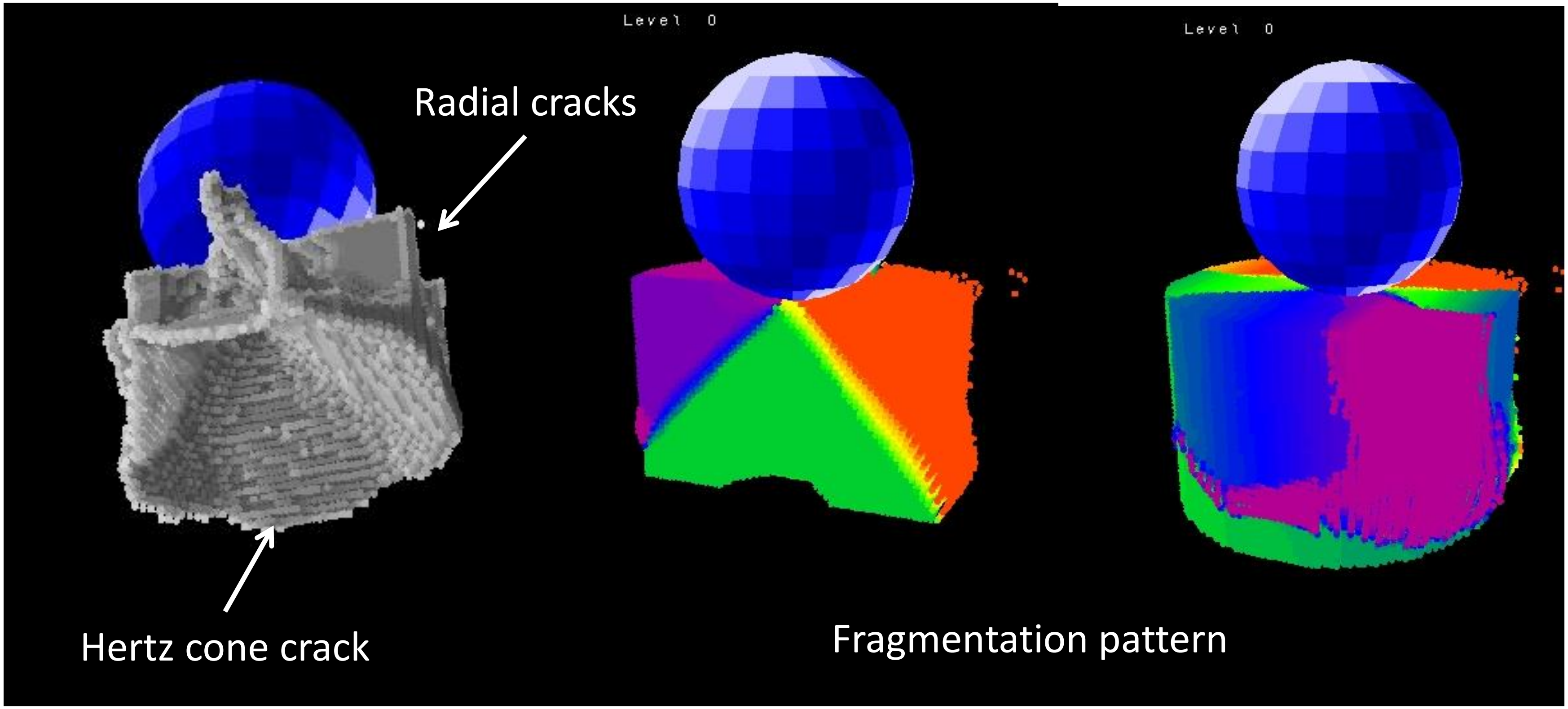


Level	Wall clock time (min) with 28K nodes in coarse grid	Wall clock time (min) with 110K nodes in coarse grid
0	30	168
2	8	16

Contact mechanics: Rigid spherical indenter



Spherical indenter, ctd.



Multiscale method discussion

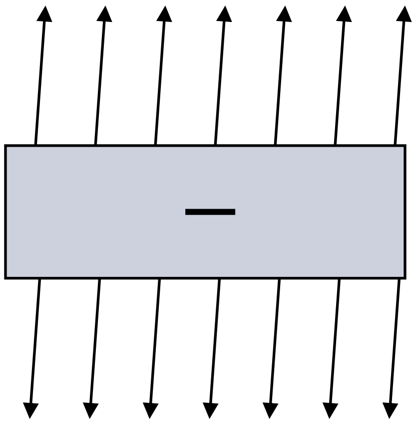
- Advantages
  - Avoids need for strong coupling (forces acting between different levels).
  - Combines multiscale with adaptive refinement.
  - Provides damaged material properties to higher levels.

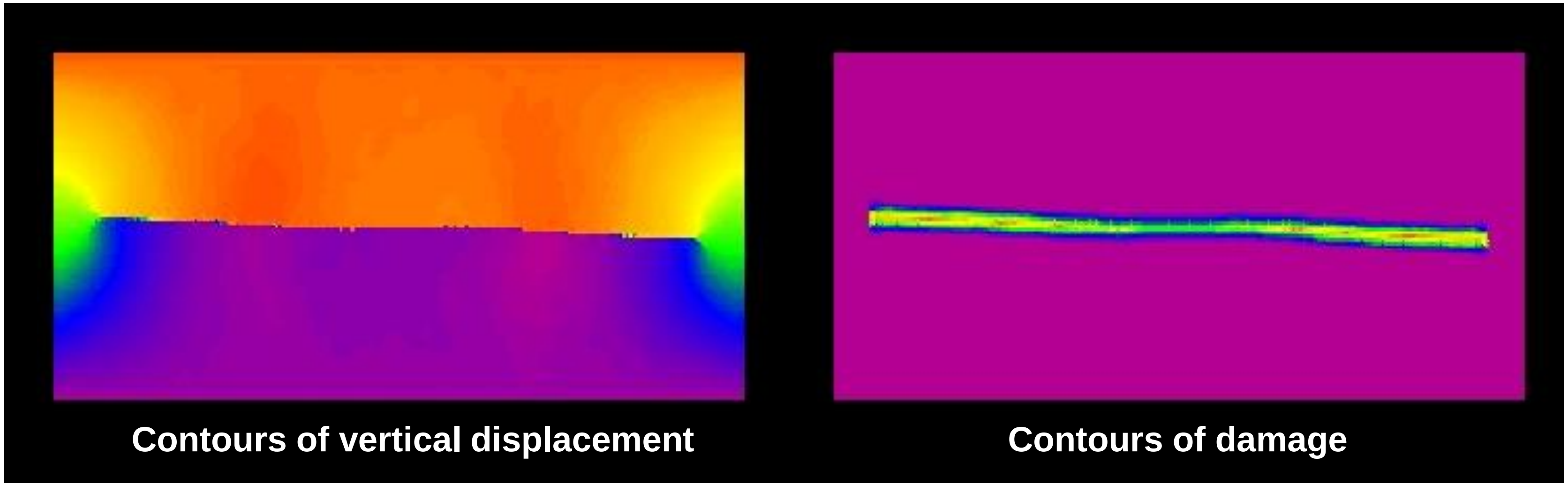
- Disadvantages
  - Difficult to know where to unrefine.
  - Pervasive fracture leads to a large number of level 0 DOFs.
  - Don’t yet have a general coarse graining method for heterogenous media.

Reduced mesh effects

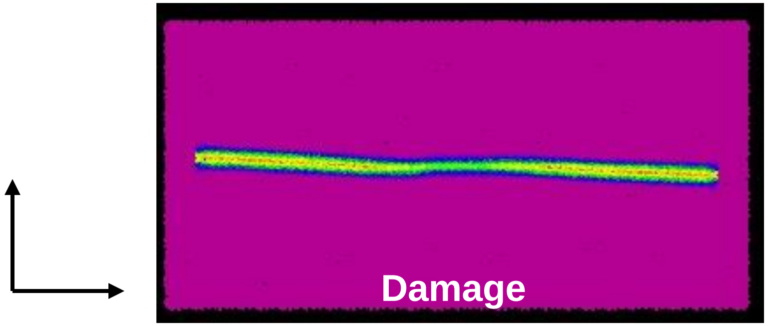
- Plate with a pre-existing defect is subjected to prescribed boundary velocities.
- These BC correspond to mostly Mode-I loading with a little Mode-II.

$$\dot{\varepsilon} = (0.25\mathrm{s}^{-1}) \begin{bmatrix} 0 & 0.1 \\ 0 & 1 \end{bmatrix}$$

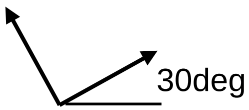




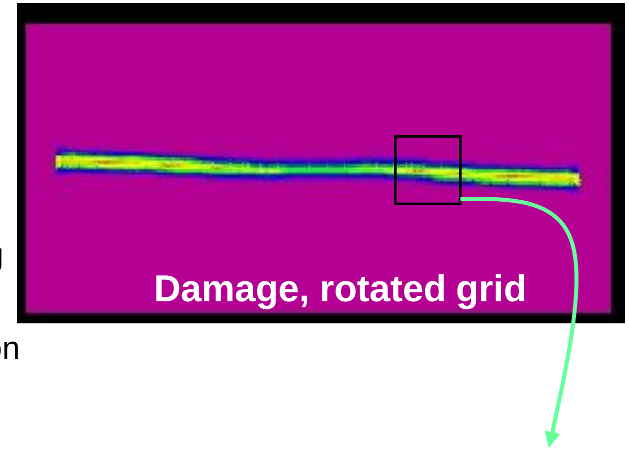
Effect of rotating the grid



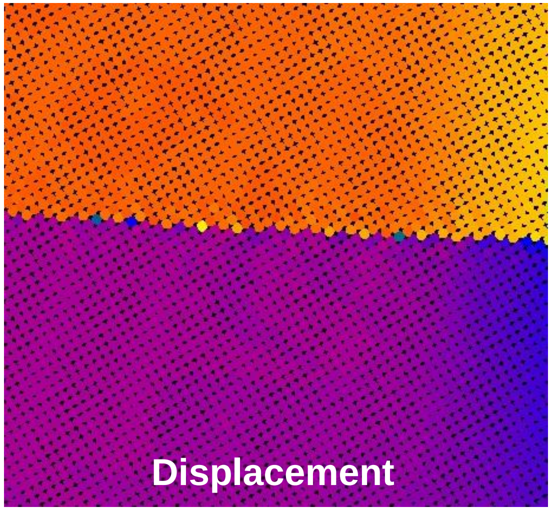
Original grid direction

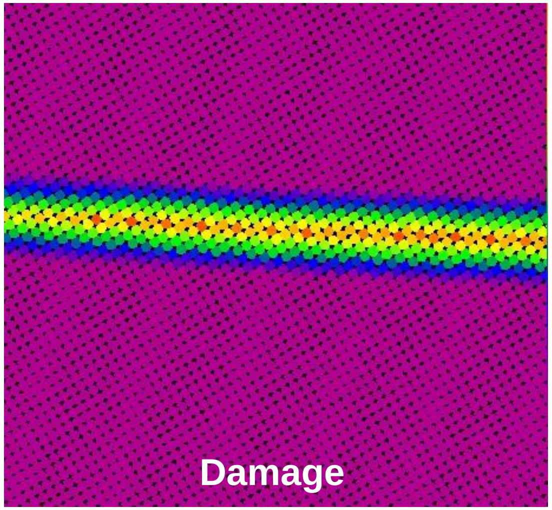


Rotated grid direction

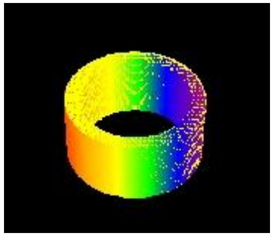


Network of identical bonds in many directions allows cracks to grow in any direction.

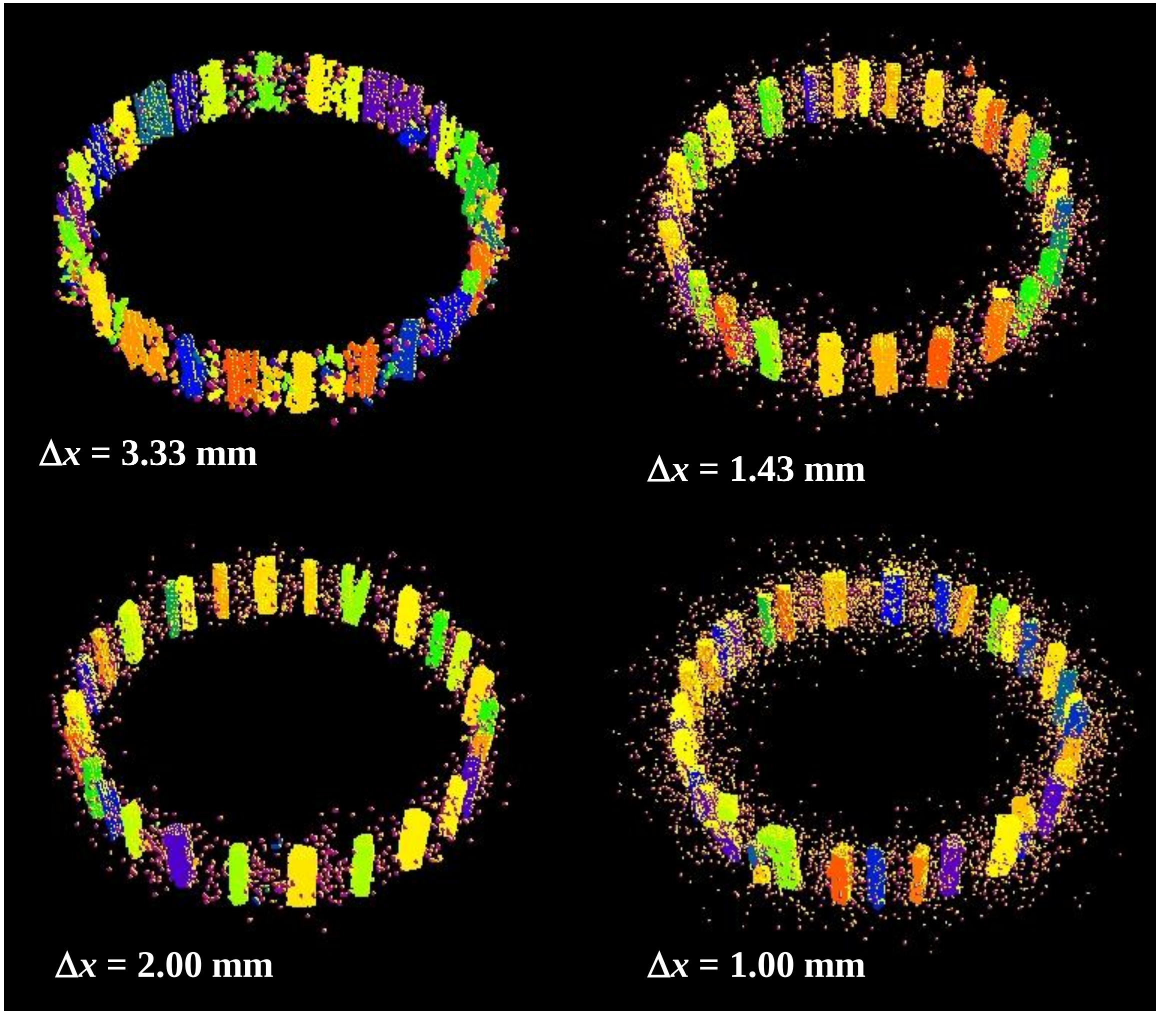




Convergence in a fragmentation problem



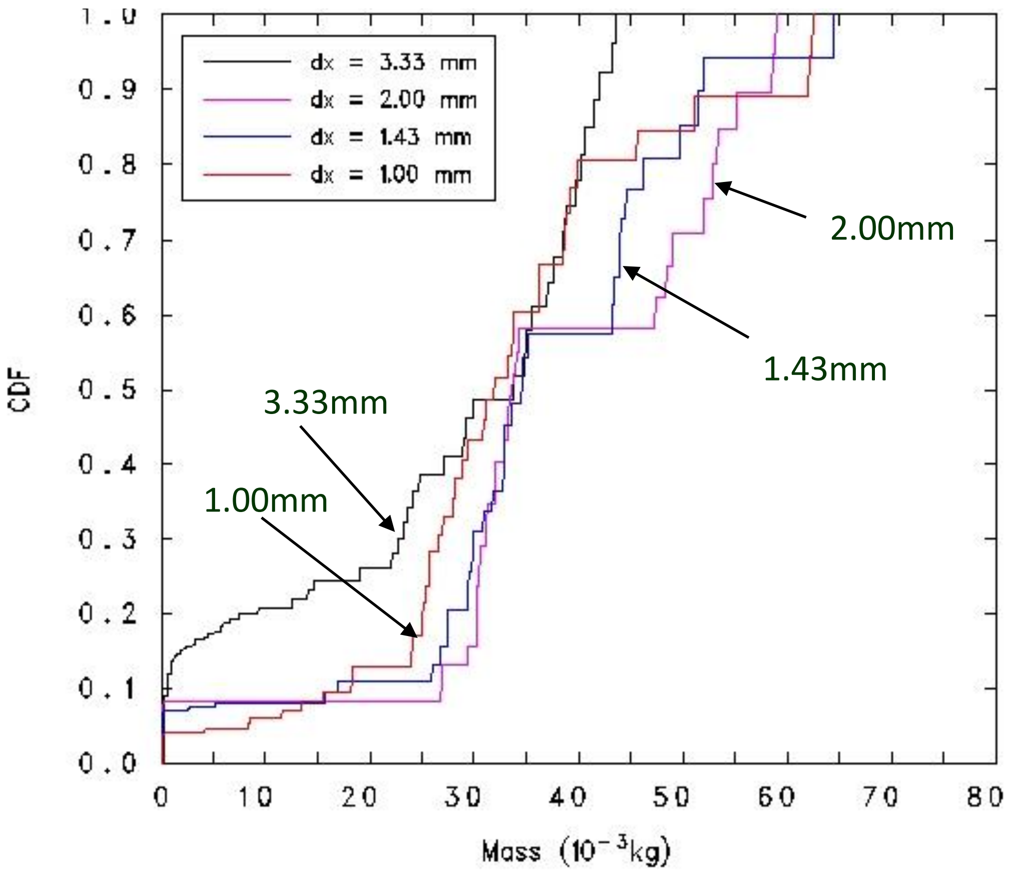
Brittle ring with initial radial velocity



$$\delta = 3\Delta x$$

Convergence in a fragmentation problem

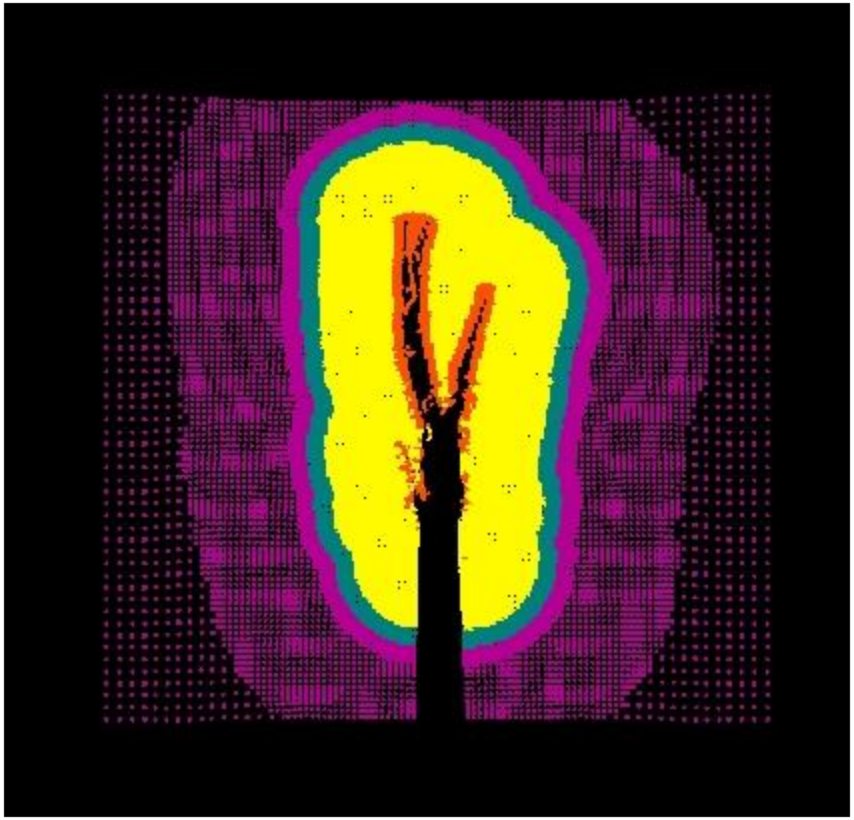
Cumulative distribution function for 4 grid spacings



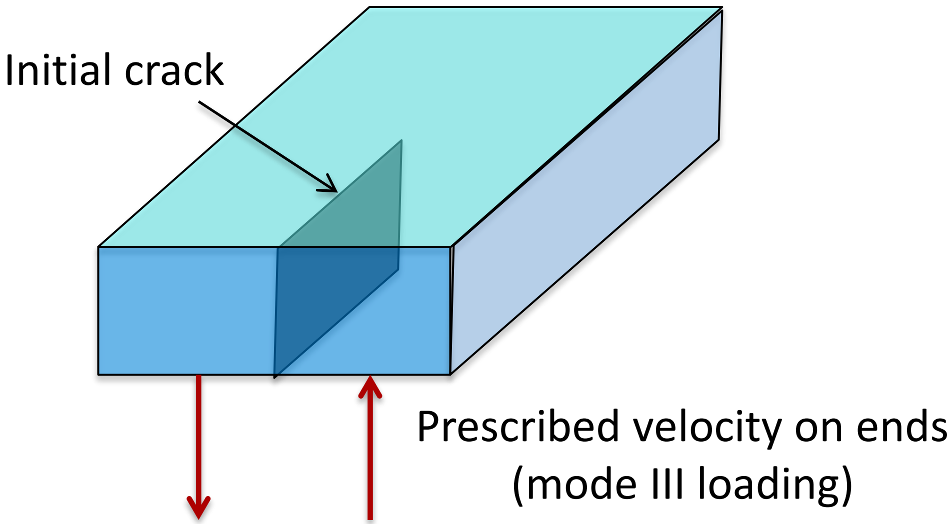
Δx (mm)	Mean fragment mass (g)
3.33	27.1
2.00	37.8
1.43	35.9
1.00	33.5

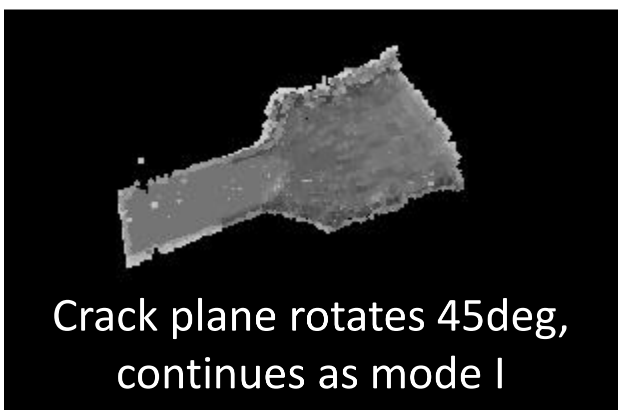
Solution appears essentially converged

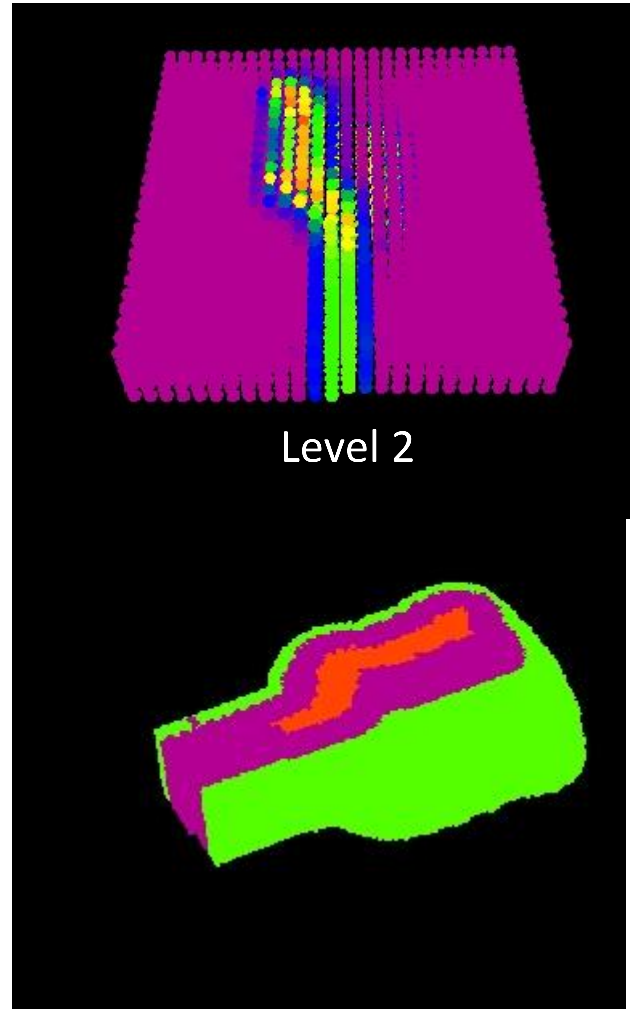
Dynamic fracture



Fracture mode transition

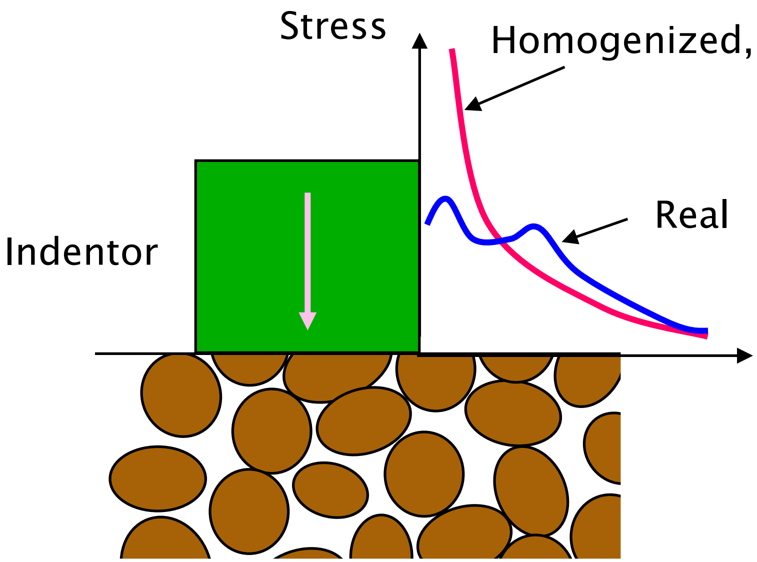



Crack plane rotates 45deg,
continues as mode I



Nonlocality as a result of homogenization

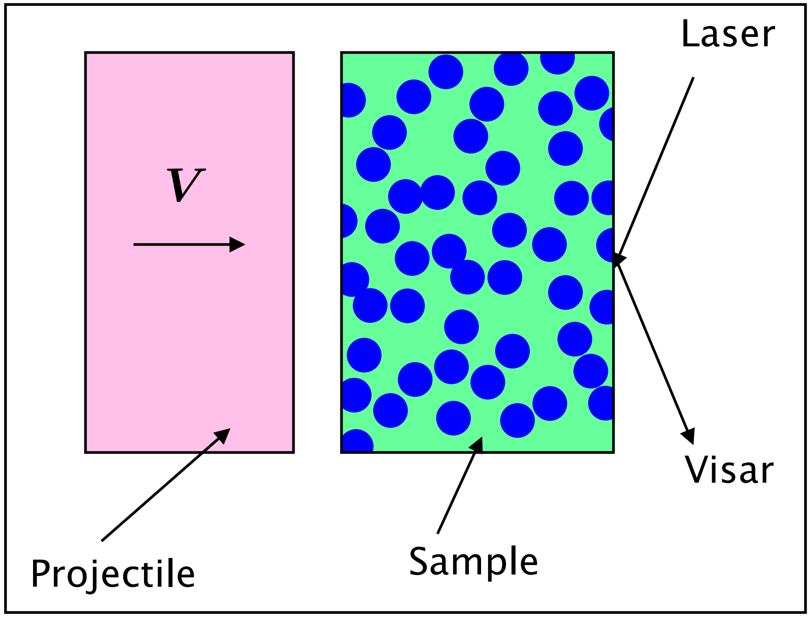
- Homogenization, neglecting the natural length scales of a system, often doesn't give good answers.

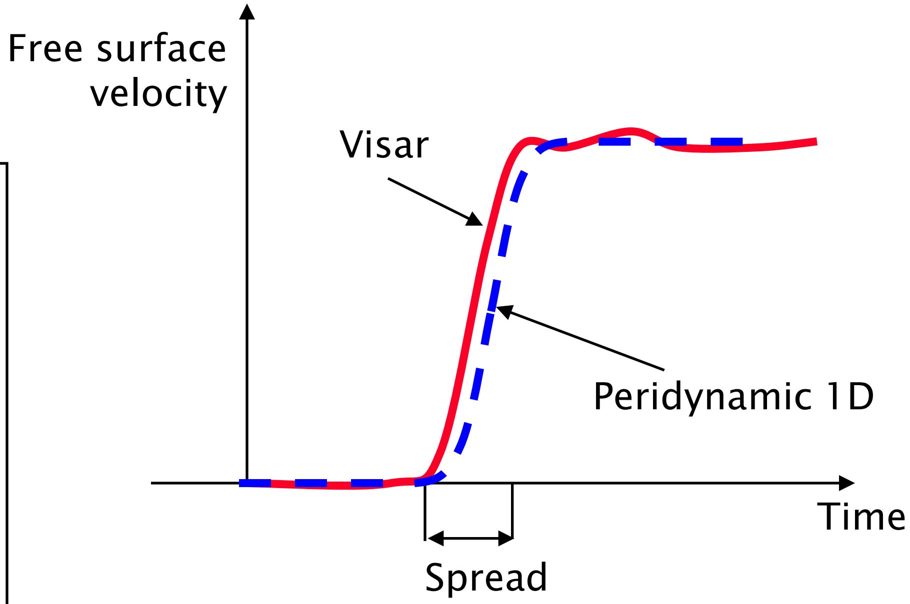


Claim: Nonlocality is an essential feature of a realistic homogenized model of a heterogeneous material.

Proposed experimental method for measuring the peridynamic horizon

- Measure how much a step wave spreads as it goes through a sample.
- Fit the horizon in a 1D peridynamic model to match the observed spread.

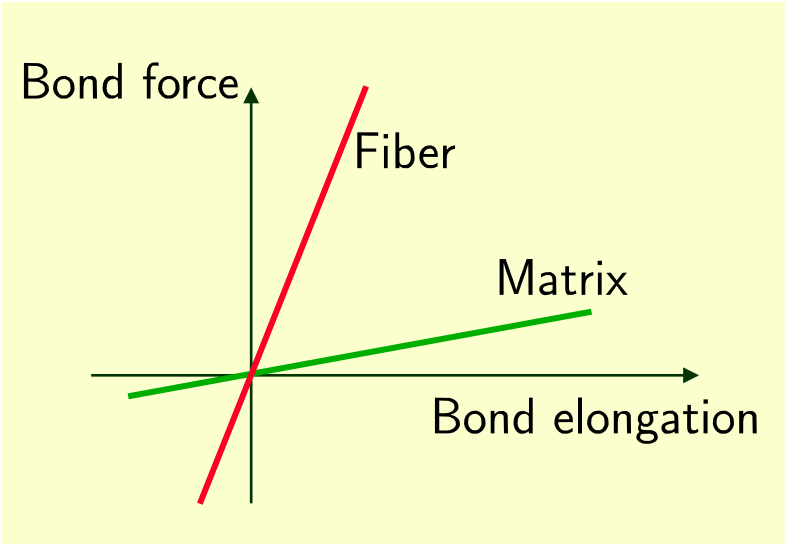


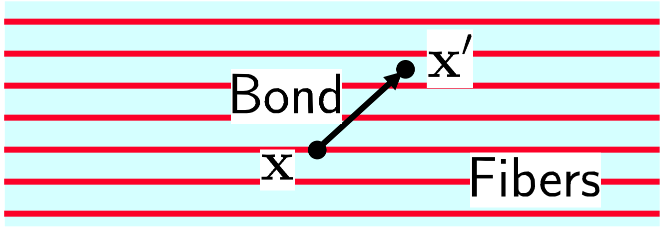


Local model would predict zero spread.

Material modeling: Composites

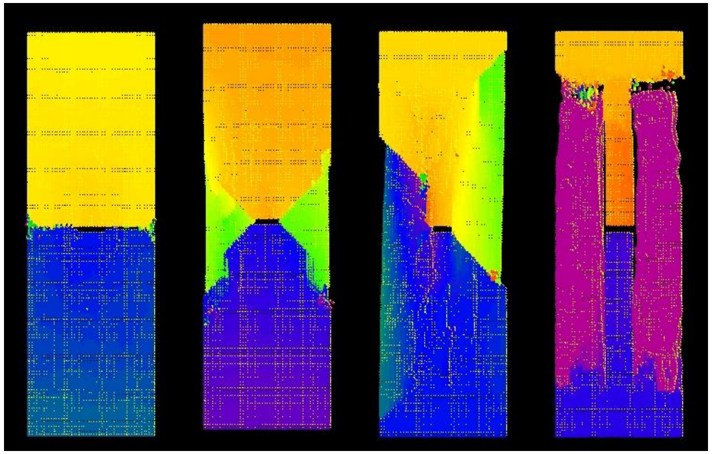
- Special case: fiber reinforced composite lamina.
- Bonds in the fiber direction are stiffer than the others.





Splitting and fracture mode change in composites

- Distribution of fiber directions between plies strongly influences the way cracks grow.



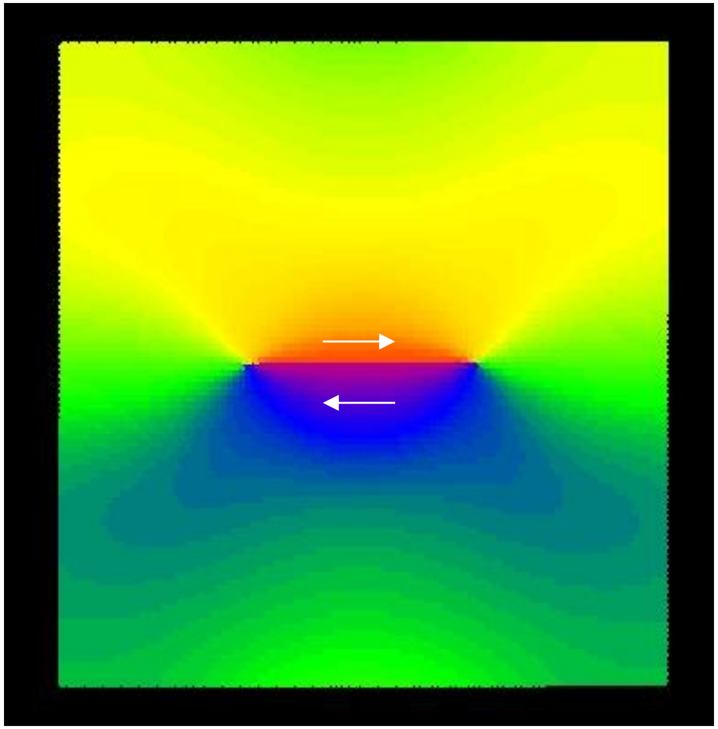
EMU simulations for different layups

Typical crack growth in a notched laminate
(photo courtesy Boeing)

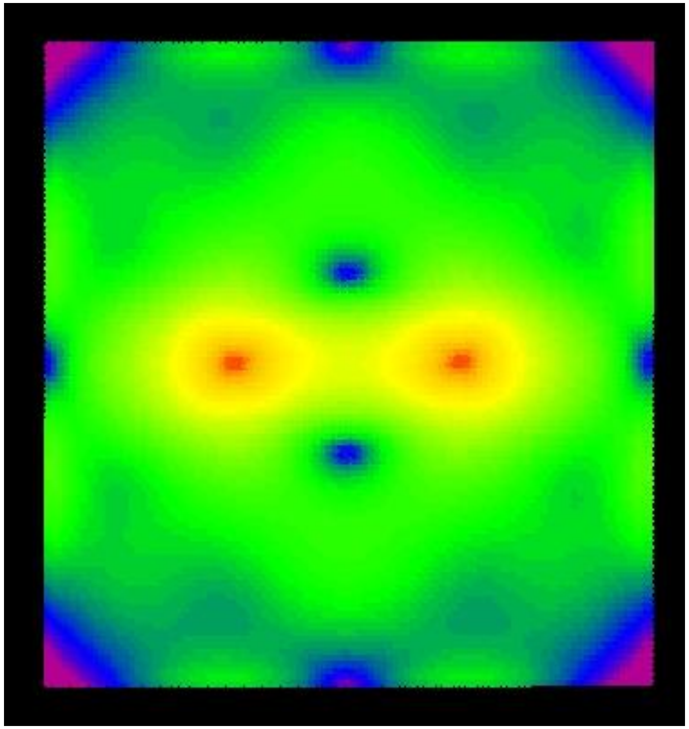
Sandia National Laboratories

Peridynamic dislocation model

Example: Dislocation segment in a square with free edges
100 x 100 EMU grid



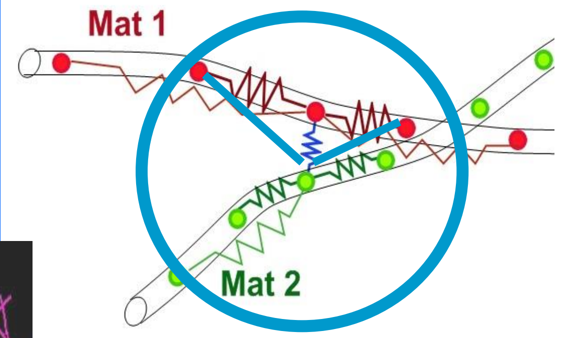
Contours of u_1



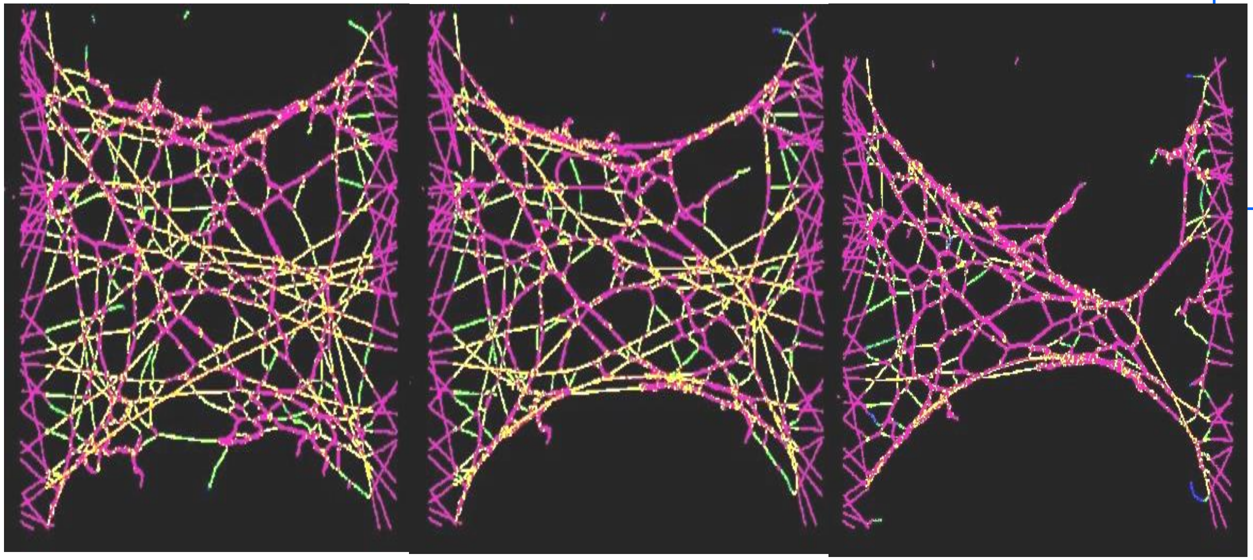
Contours of $\log W$
W=elastic energy density

Example of long-range forces: Nanofiber network

- Peridynamics treats all internal forces as long-range.
- This makes it a natural way to treat van der Waals and surface forces.



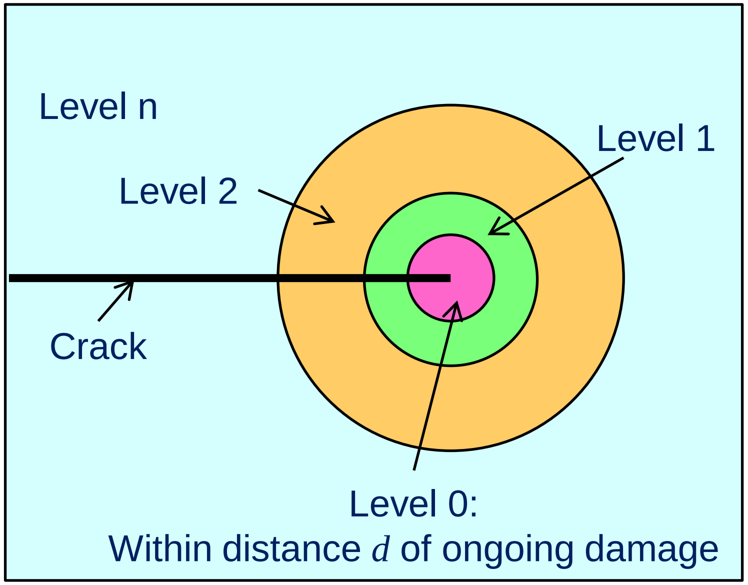
Nanofiber interactions due to van der Waals forces



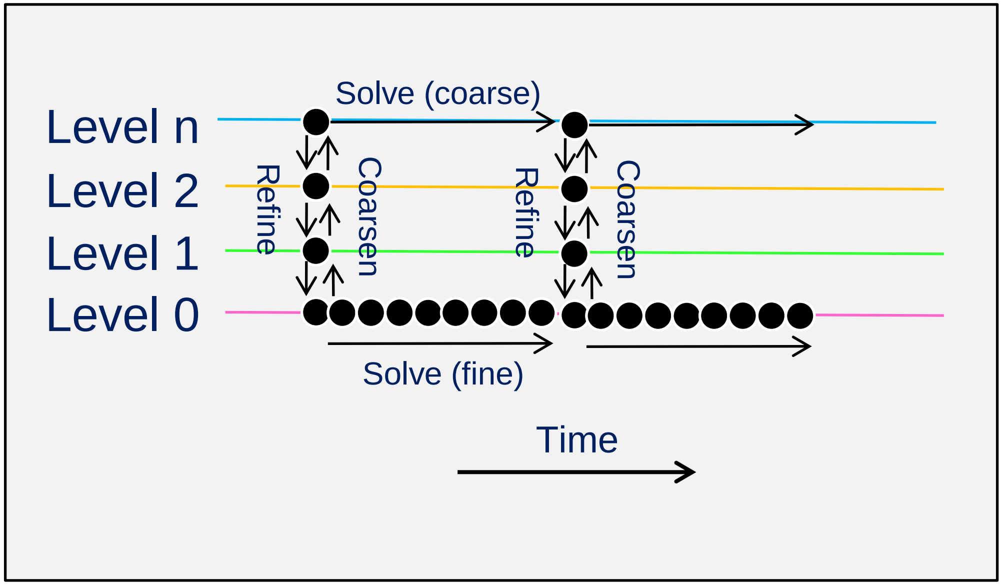
Nanofiber membrane (F. Bobaru, Univ. of Nebraska)

Concurrent solution strategy

Level 0 region follows the crack tip



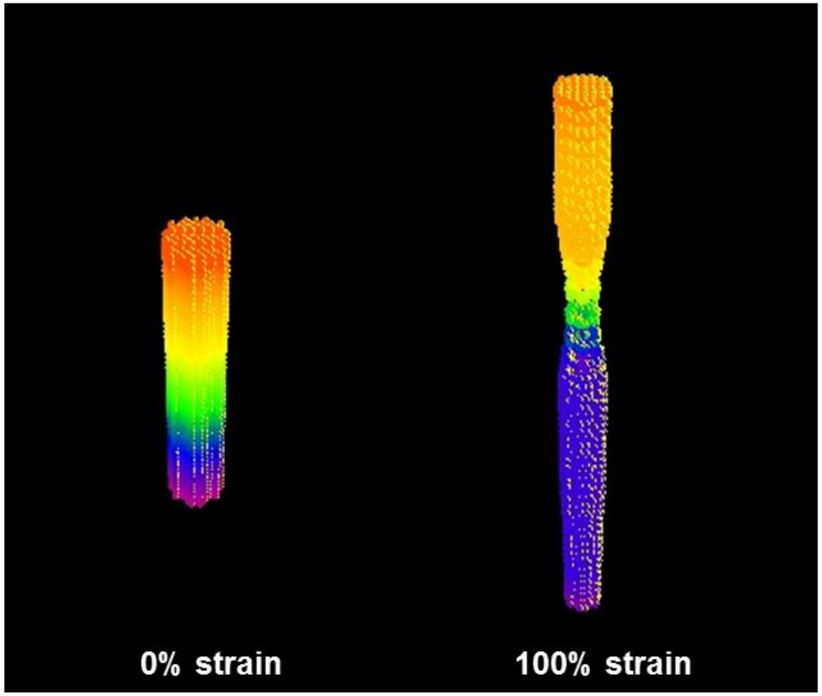
Concurrent solution strategy



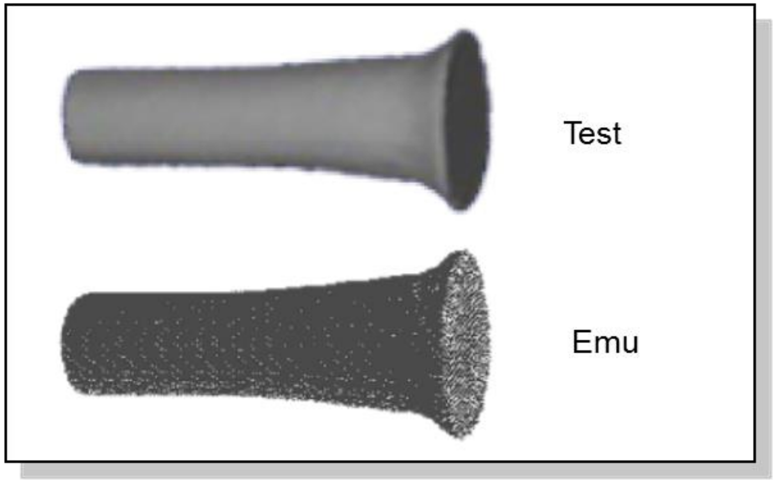
- Refinement:
  - Level 1 acts as a boundary condition on level 0.
- Coarsening:
  - Level 0 supplies material properties (e.g., damage) to higher levels.

Any standard material model can be used in peridynamics

- Example: Large-deformation, strain-hardening, rate-dependent material model.
  - Material model implementation by John Foster.



Necking of a bar under tension



Taylor impact test

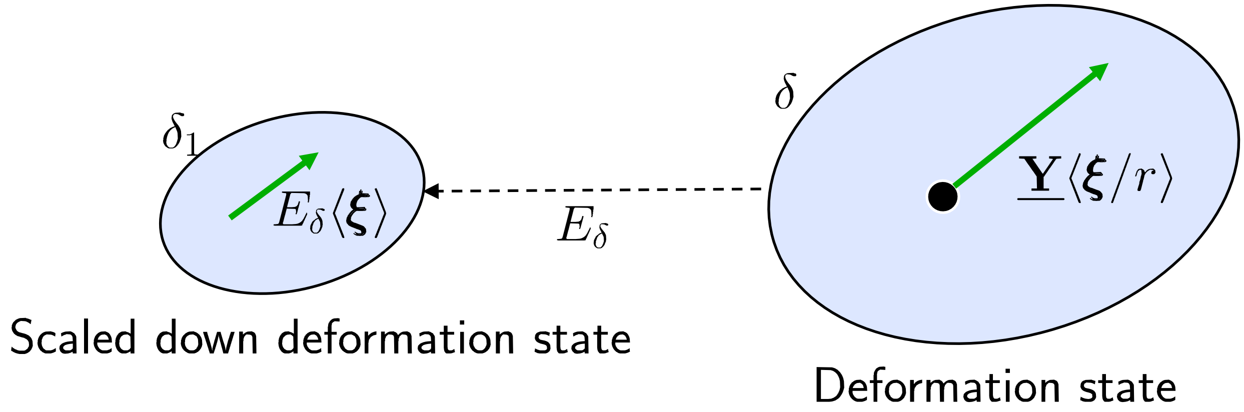
Rescaling an elastic material model

- Start with a material model W_1 which has some fixed horizon δ_1.
- Define a mapping that takes a new, larger horizon δ into the original:

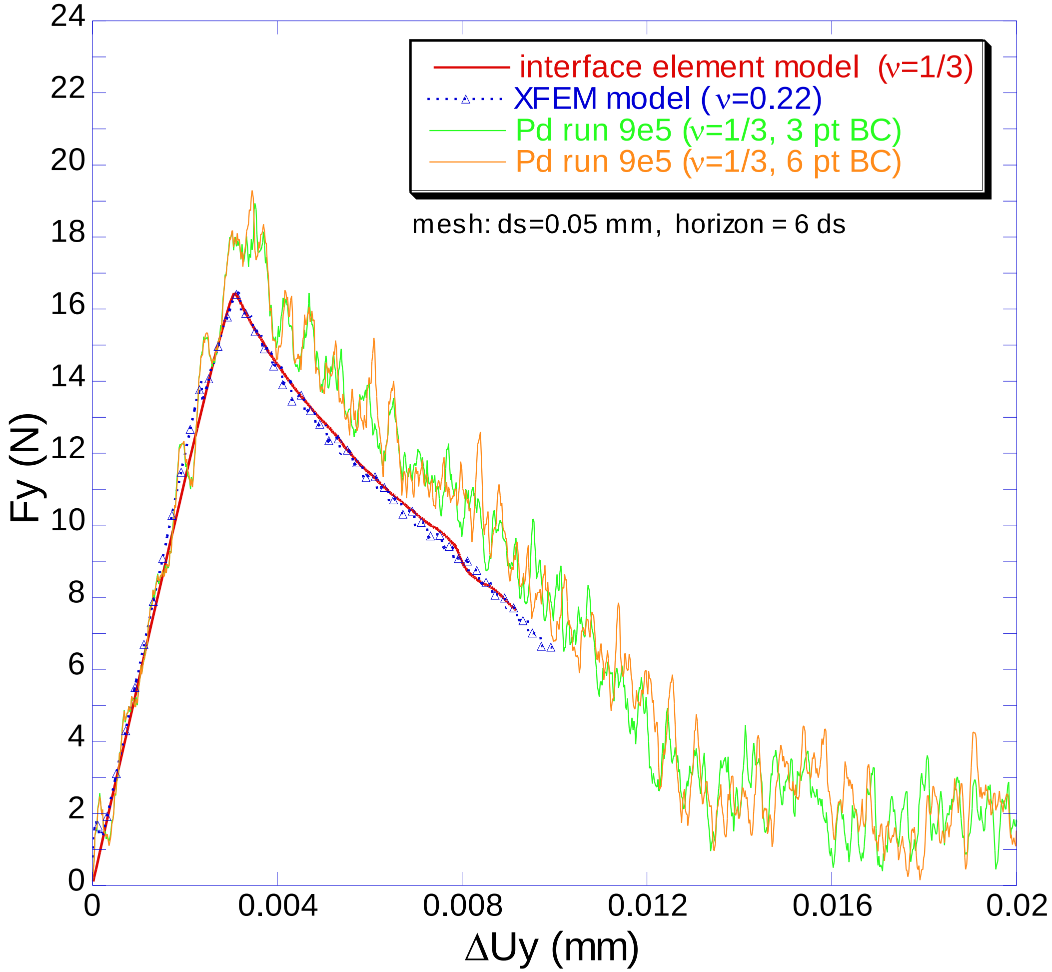
$$(E_\delta(\underline{\mathbf{Y}}))\langle\boldsymbol{\xi}\rangle = r\underline{\mathbf{Y}}\langle\boldsymbol{\xi}/r\rangle, \qquad r = \frac{\delta_1}{\delta} \leq 1$$

- Then set

$$W_\delta(\underline{\mathbf{Y}}) = W_1(E_\delta(\underline{\mathbf{Y}}))$$



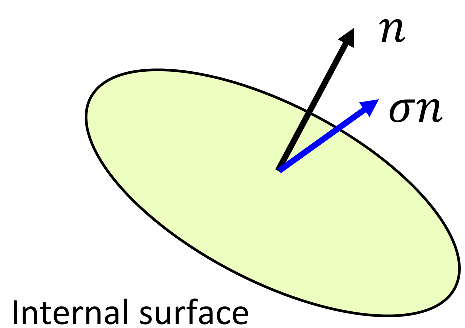
Comparison with XFEM, interface elements



Peridynamics basics:
The nature of internal forces

Standard theory

Stress tensor field
(assumes contact forces and
smooth deformation)

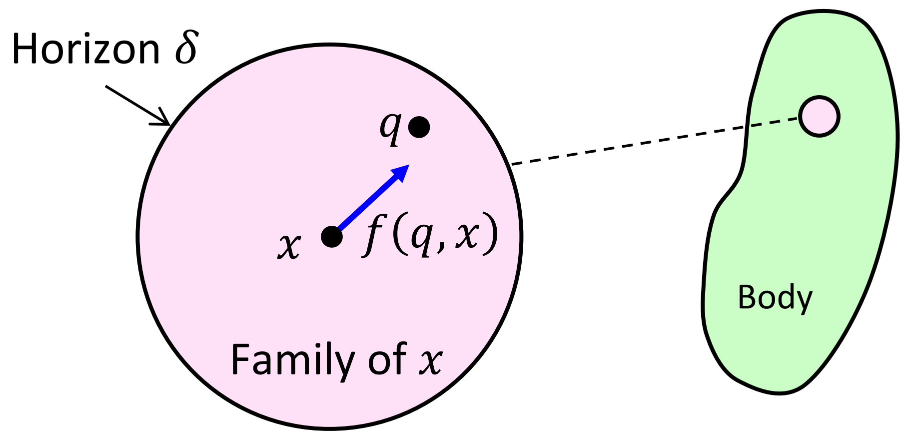


$$\rho\ddot{u}(x,t) = \nabla \cdot \sigma(x,t) + b(x,t)$$

Differentiation of contact forces

Peridynamics

Bond forces within small neighborhoods
(allow discontinuity)



$$\rho\ddot{u}(x,t) = \int_{H_x} f(q,x)dV_q + b(x,t)$$

Summation over bond forces

Peridynamics basics: States

- A *peridynamic state* is a mapping on bonds in a family.
- We write:

$$\mathbf{u} = \underline{\mathbf{A}}\langle \boldsymbol{\xi} \rangle$$

  where $\boldsymbol{\xi}$ is a bond, $\underline{\mathbf{A}}$ is a state, and $\mathbf{u}$ is some vector.
- States play a role in peridynamics similar to that of second order tensors in the local theory.

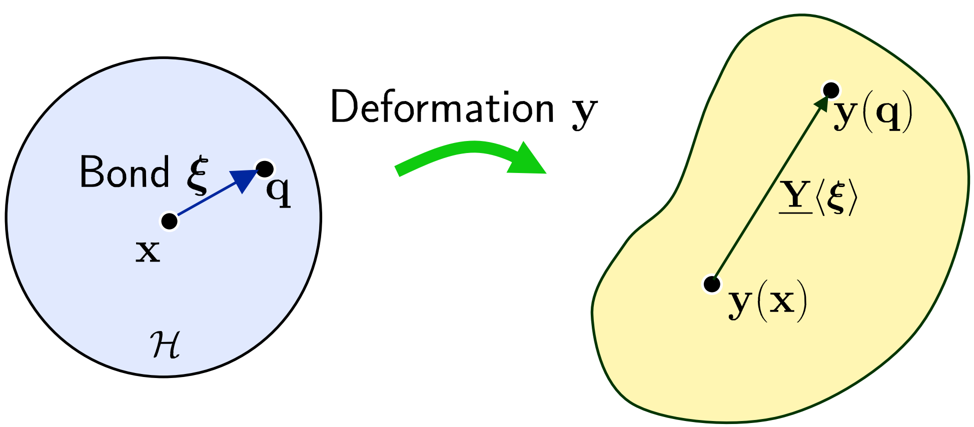
Peridynamics basics: Kinematics

- The *deformation state* is the function that maps each bond $\boldsymbol{\xi}$ into its deformed image:

$$\underline{\mathbf{Y}}\langle\boldsymbol{\xi}\rangle = \mathbf{y}(\mathbf{q}) - \mathbf{y}(\mathbf{x})$$

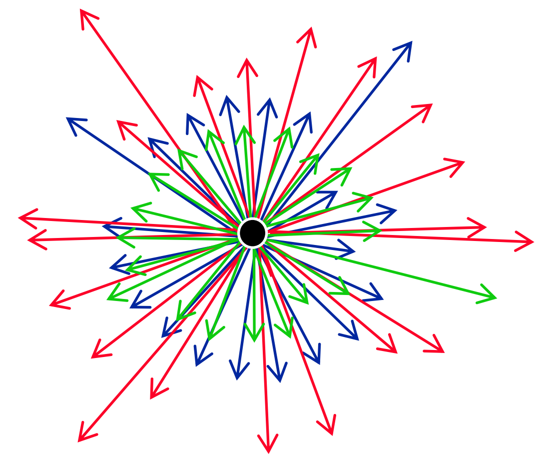
where $\mathbf{y}$ is the deformation and

$$\boldsymbol{\xi} = \mathbf{q} - \mathbf{x}.$$



Undeformed family of $\mathbf{x}$

Deformed family of $\mathbf{x}$



Deformed images of bonds:
State description allows complexity

Peridynamics basics: Force state

- $\mathbf{f}(\mathbf{x},\mathbf{q})$ has contributions from the material models at both $\mathbf{x}$ and $\mathbf{q}$.

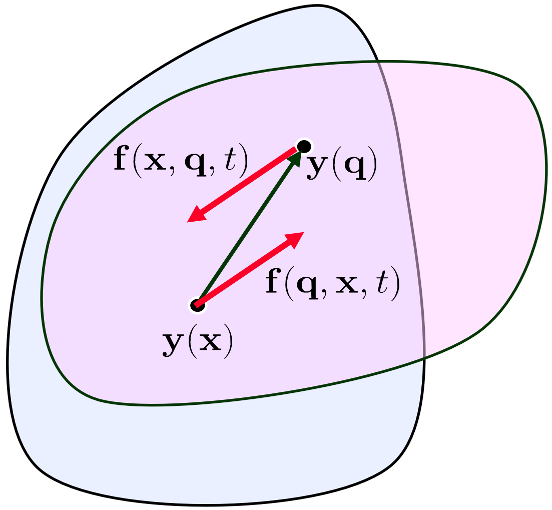
$$\mathbf{f}(\mathbf{x},\mathbf{q}) = \mathbf{t}(\mathbf{x},\mathbf{q}) - \mathbf{t}(\mathbf{q},\mathbf{x})$$

$$\mathbf{t}(\mathbf{x},\mathbf{q}) = \underline{\mathbf{T}}[\mathbf{x}]\langle \mathbf{q}-\mathbf{x}\rangle, \qquad \mathbf{t}(\mathbf{x},\mathbf{q}) = \underline{\mathbf{T}}[\mathbf{q}]\langle \mathbf{x}-\mathbf{q}\rangle$$

- $\underline{\mathbf{T}}[\mathbf{x}]$ is the *force state*: maps bonds onto bond force densities. It is found from the constitutive model:

$$\underline{\mathbf{T}} = \hat{\underline{\mathbf{T}}}(\underline{\mathbf{Y}})$$

where $\hat{\underline{\mathbf{T}}}$ maps the deformation state to the force state.



Peridynamics basics: Elastic materials

- A peridynamic elastic material has strain energy density given by

$$W(\underline{\mathbf{Y}}).$$

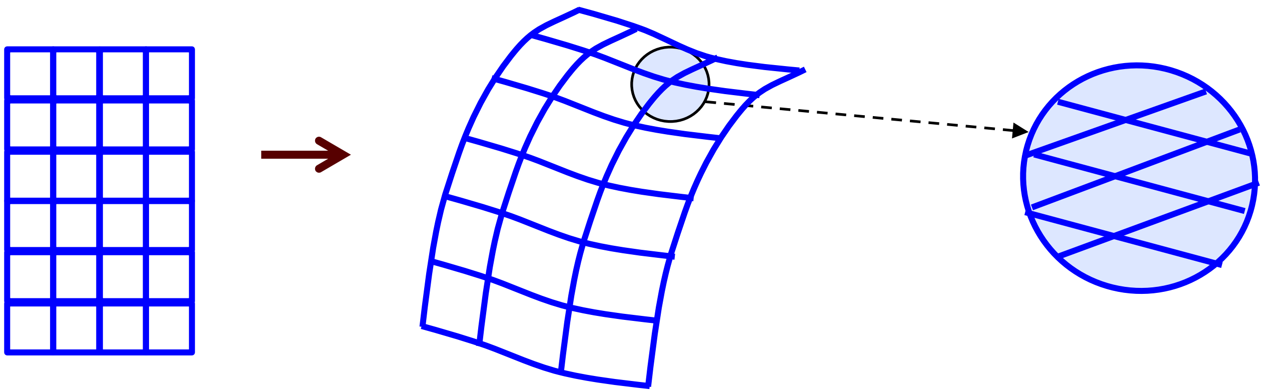
- The force state is given by

$$\hat{\underline{\mathbf{T}}}(\underline{\mathbf{Y}}) = W_{\underline{\mathbf{Y}}}(\underline{\mathbf{Y}})$$

where $W_{\underline{\mathbf{Y}}}$ is the Frechet derivative of the strain energy density.

Peridynamics converges to the local theory

- Can prove that if the deformation is smooth, then in the limit $\delta \to 0$ while holding the bulk material properties constant, for any bond $\boldsymbol{\xi}$:
- $\underline{\mathbf{Y}}\langle\boldsymbol{\xi}\rangle \to \mathbf{F}\boldsymbol{\xi}$, where $\mathbf{F}$=deformation gradient tensor
- There exists a tensor field $\boldsymbol{\sigma}$ such that $\int \mathbf{f} \to \nabla \cdot \boldsymbol{\sigma}$, so the standard PDE is recovered.



In this sense, the standard theory is a subset of peridynamics.

*Joint work with R. Lehoucq

Some results about peridynamics

- For any choice of horizon, we can fit material model parameters to match the bulk properties and energy release rate.
  - Using nonlocality, can obtain material model parameters from wave dispersion curves (Weckner).
- Coupled coarse scale and fine scale evolution equations can be derived for composites (Lipton and Alali).
- A set of discrete particles interacting through any multibody potential can be represented exactly as a peridynamic body.
- Well posedness has been established under certain conditions (Mangesha, Du, Gunzburger, Lehoucq).

EMU numerical method

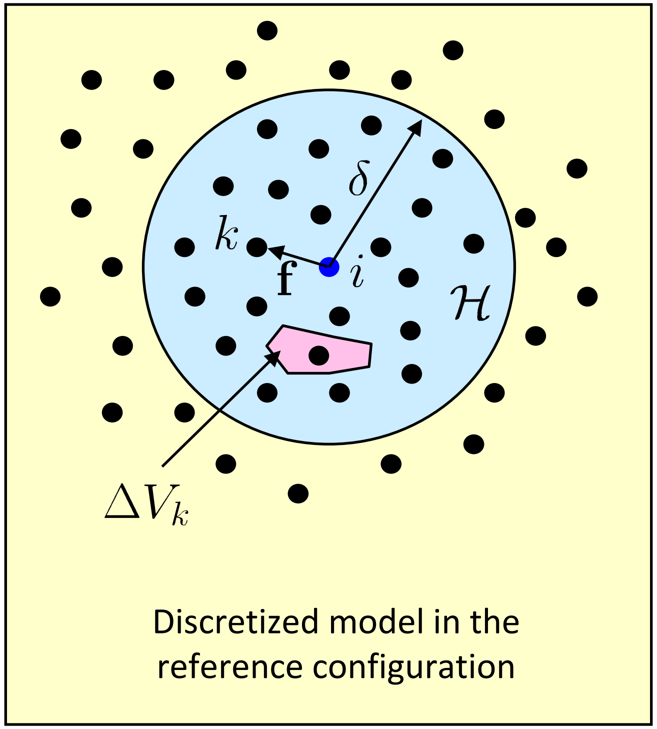
- Integral is replaced by a finite sum: resulting method is meshless and Lagrangian.

$$\rho \ddot{\mathbf{y}}(\mathbf{x}, t) = \int_{\mathcal{H}} \mathbf{f}(\mathbf{q}, \mathbf{x}, t)\, dV_{\mathbf{q}} + \mathbf{b}(\mathbf{x}, t)$$

$$\downarrow$$

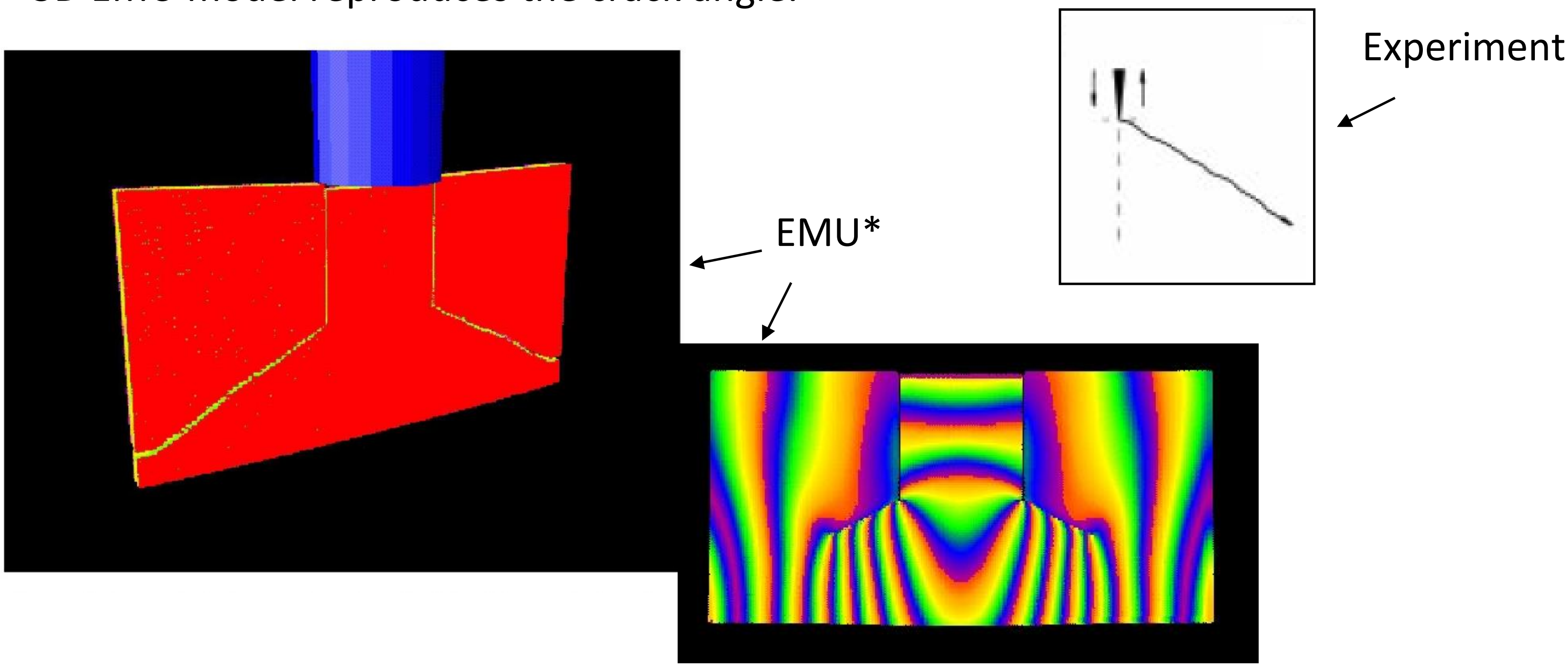
$$\rho \ddot{\mathbf{y}}_i^n = \sum_{k \in \mathcal{H}} \mathbf{f}(\mathbf{x}_k, \mathbf{x}_i, t)\, \Delta V_k + \mathbf{b}_i^n$$

- Looks a lot like MD.
- Unrelated to Smoothed Particle Hydrodynamics
    - SPH solves the local equations by fitting spatial derivatives to the current node values.



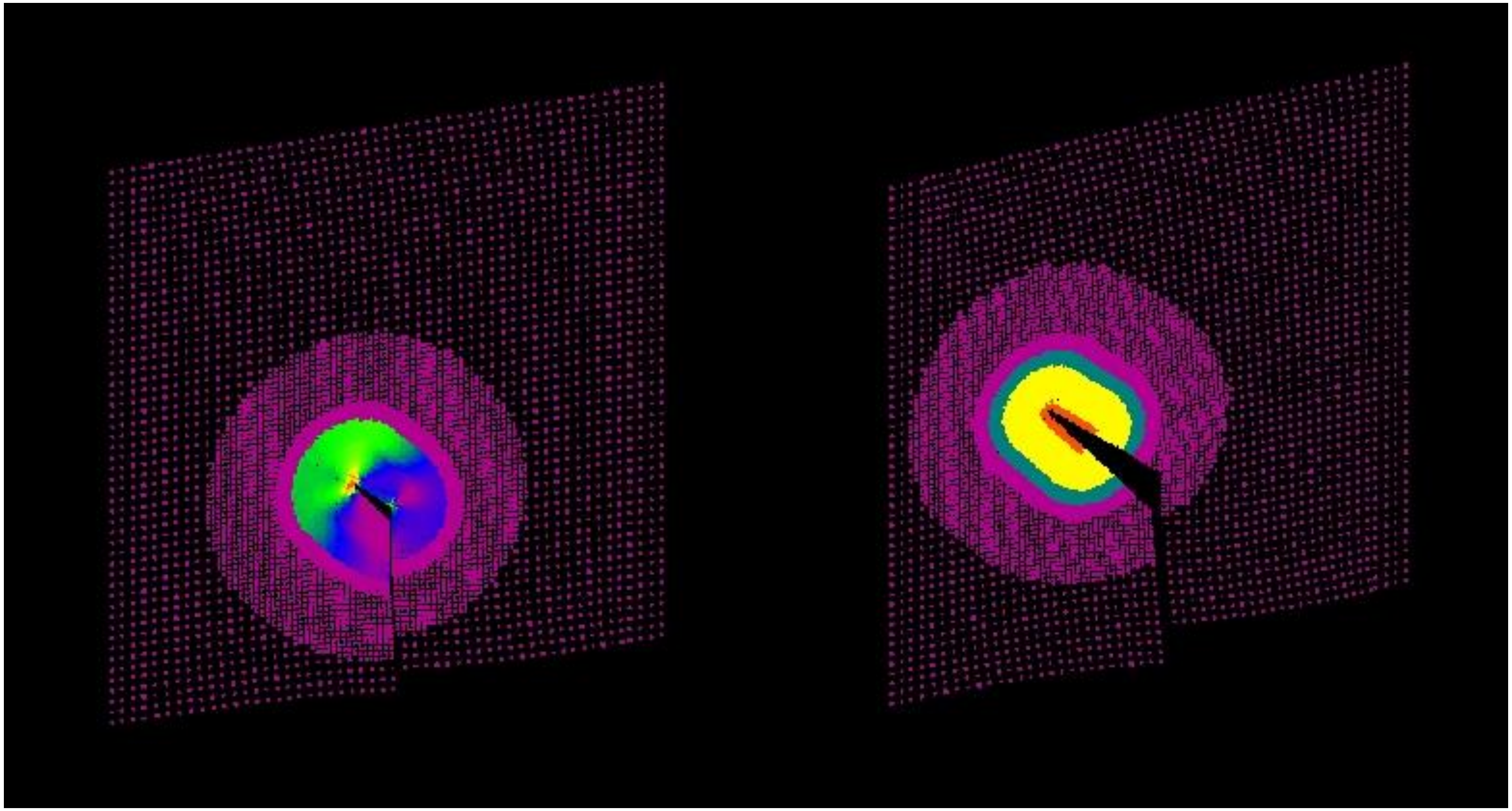
Example: Dynamic fracture

- Dynamic fracture in maraging steel (Kalthoff & Winkler, 1988)
- Mode-II loading at notch tips results in mode-I cracks at 70deg angle.
- 3D EMU model reproduces the crack angle.



S. A. Silling, Dynamic fracture modeling with a meshfree peridynamic code, in *Computational Fluid and Solid Mechanics 2003*, K.J. Bathe, ed., Elsevier, pp. 641-644.

Shear loading

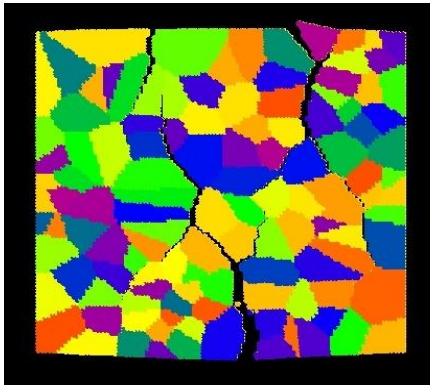


Bond strain

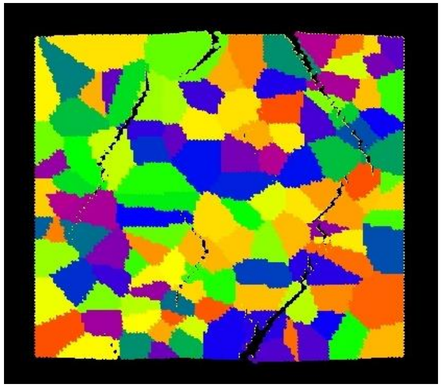
Damage process zone

Polycrystals: Mesoscale model*

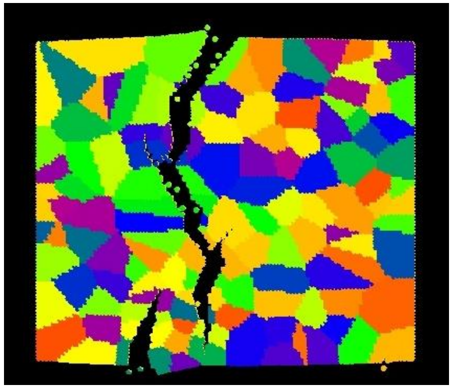
- What is the effect of grain boundaries on the fracture of a polycrystal?



$\beta = 0.25$



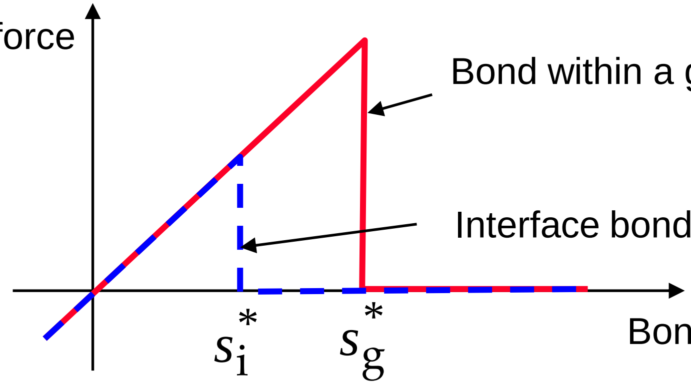
$\beta = 1$



$\beta = 4$

$$\beta = \frac{s_i^*}{s_g^*}$$

Large β favors trans-granular fracture.



* Work by F. Bobaru & students

Peridynamic vs. local equations

State notation: $\underline{\text{State}}\langle\text{bond}\rangle = \text{vector}$

Relation	*Peridynamic theory*	*Standard theory*
Kinematics	$\underline{\mathbf{Y}}\langle\mathbf{q}-\mathbf{x}\rangle = \mathbf{y}(\mathbf{q}) - \mathbf{y}(\mathbf{x})$	$\mathbf{F}(\mathbf{x}) = \frac{\partial\mathbf{y}}{\partial\mathbf{x}}(\mathbf{x})$
Linear momentum balance	$\rho\ddot{\mathbf{y}}(\mathbf{x}) = \int_{\mathcal{H}} \Big(\mathbf{t}(\mathbf{q},\mathbf{x}) - \mathbf{t}(\mathbf{x},\mathbf{q})\Big)\, dV_{\mathbf{q}} + \mathbf{b}(\mathbf{x})$	$\rho\ddot{\mathbf{y}}(\mathbf{x}) = \nabla\cdot\boldsymbol{\sigma}(\mathbf{x}) + \mathbf{b}(\mathbf{x})$
Constitutive model	$\mathbf{t}(\mathbf{q},\mathbf{x}) = \underline{\mathbf{T}}\langle\mathbf{q}-\mathbf{x}\rangle, \qquad \underline{\mathbf{T}} = \hat{\underline{\mathbf{T}}}(\underline{\mathbf{Y}})$	$\boldsymbol{\sigma} = \hat{\boldsymbol{\sigma}}(\mathbf{F})$
Angular momentum balance	$\int_{\mathcal{H}} \underline{\mathbf{Y}}\langle\mathbf{q}-\mathbf{x}\rangle \times \underline{\mathbf{T}}\langle\mathbf{q}-\mathbf{x}\rangle\, dV_{\mathbf{q}} = \mathbf{0}$	$\boldsymbol{\sigma} = \boldsymbol{\sigma}^T$
Elasticity	$\underline{\mathbf{T}} = W_{\underline{\mathbf{Y}}}$ (Fréchet derivative)	$\boldsymbol{\sigma} = W_{\mathbf{F}}$ (tensor gradient)
First law	$\dot{\varepsilon} = \underline{\mathbf{T}} \bullet \dot{\underline{\mathbf{Y}}} + q + r$	$\dot{\varepsilon} = \boldsymbol{\sigma}\cdot\dot{\mathbf{F}} + q + r$

$$\underline{\mathbf{T}} \bullet \dot{\underline{\mathbf{Y}}} := \int_{\mathcal{H}} \underline{\mathbf{T}}\langle\boldsymbol{\xi}\rangle \cdot \dot{\underline{\mathbf{Y}}}\langle\boldsymbol{\xi}\rangle\, dV_{\boldsymbol{\xi}}$$

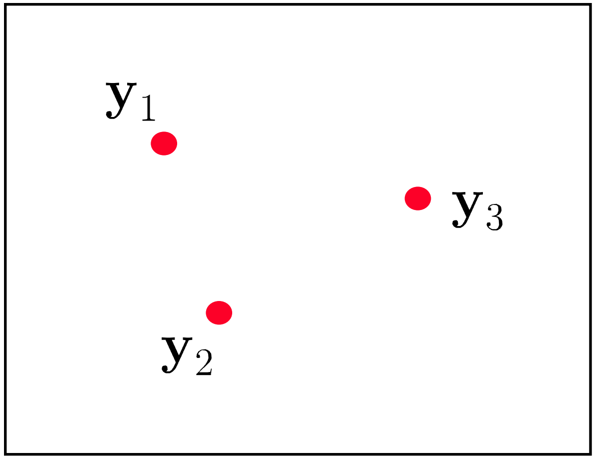
Discrete particles and PD states

- Consider a set of atoms that interact through an $N-$body potential:

$$U(\mathbf{y}_1, \mathbf{y}_2, \ldots, \mathbf{y}_N),$$

$\mathbf{y}_1, \ldots, \mathbf{y}_N$ = deformed positions, $\mathbf{x}_1, \ldots, \mathbf{x}_N$ = reference positions.

- This can be represented exactly as a peridynamic body.



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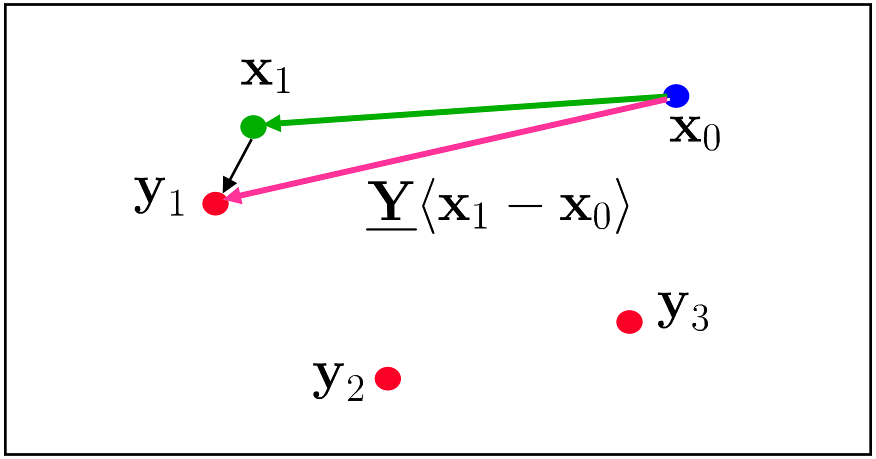
Discrete particles and PD states, ctd.

Define a peridynamic body by:

$$\hat{W}(\underline{\mathbf{Y}}, \mathbf{x}) = \Delta(\mathbf{x}-\mathbf{x}_0)U(\underline{\mathbf{Y}}\langle\mathbf{x}_1-\mathbf{x}_0\rangle, \underline{\mathbf{Y}}\langle\mathbf{x}_2-\mathbf{x}_0\rangle, \ldots, \underline{\mathbf{Y}}\langle\mathbf{x}_N-\mathbf{x}_0\rangle),$$

$$\rho(\mathbf{x}) = \sum_i \Delta(\mathbf{x}-\mathbf{x}_i)M_i$$



Discrete particles and PD states, ctd.

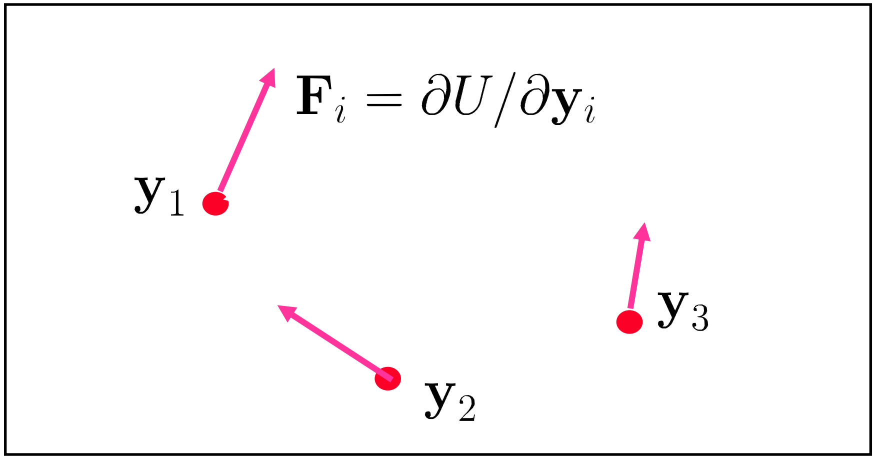
After evaluating the Frechet derivative $\underline{\mathbf{T}}$, find

$$\rho(\mathbf{x})\ddot{\mathbf{y}}(\mathbf{x},t) = \int \mathbf{f}(\mathbf{x}',\mathbf{x},t)\, dV_{\mathbf{x}'}$$

implies

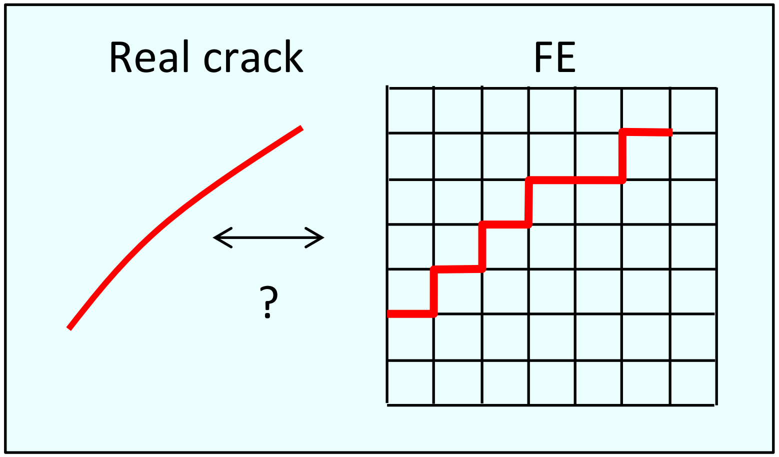
$$M_i\ddot{\mathbf{y}}(\mathbf{x}_i,t) = -\frac{\partial U}{\partial \mathbf{y}_i}, \qquad i = 1,\ldots,N$$

In other words, the PD equation of motion reduces to Newton's second law.



Why this is important

- The standard PDEs are incompatible with the essential physical nature of cracks.
  - Can’t apply PDEs on a discontinuity.
- Typical FE approaches implement a fracture model <u>after</u> numerical discretization.
  - Need supplemental kinetic relations that are understood only in idealized cases.



Complex crack path in a composite

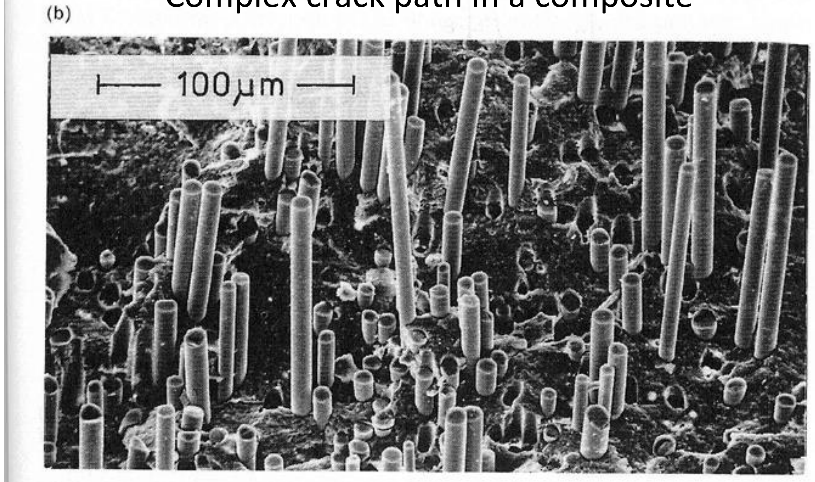


Figure 11.20 Pull-out: (a) schematic diagram; (b) fracture surface of ‘Silceram’ glass-ceramic reinforced with SiC fibres. (Courtesy H. S. Kim, P. S. Rogers and R. D. Rawlings.)