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Grain-scale Experimental Validation of Crystal Plasticity Finite Element Simulations of Tantalum Oligocrystals

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7th Multiscale Materials Modeling (MMM) International Conference
October 8, 2014. Berkeley, CA

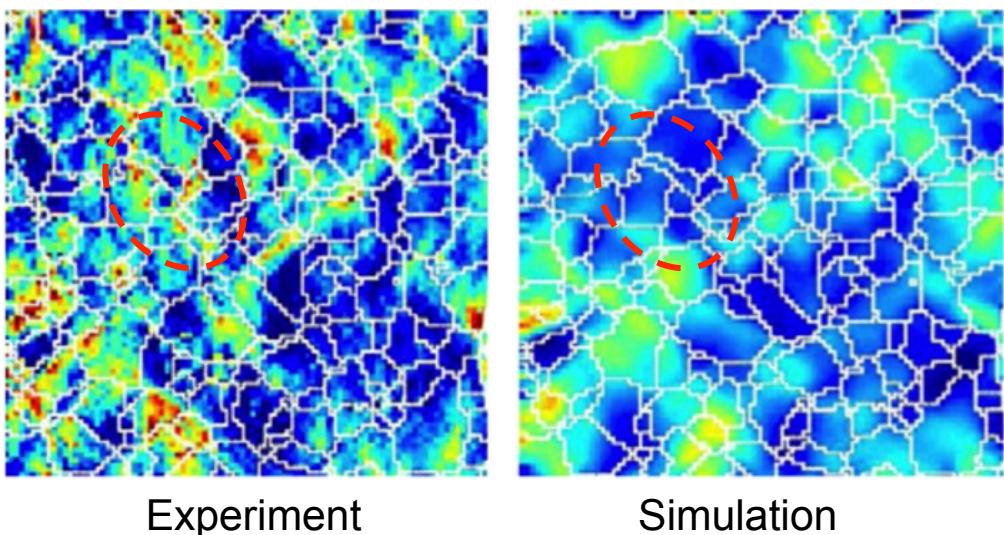


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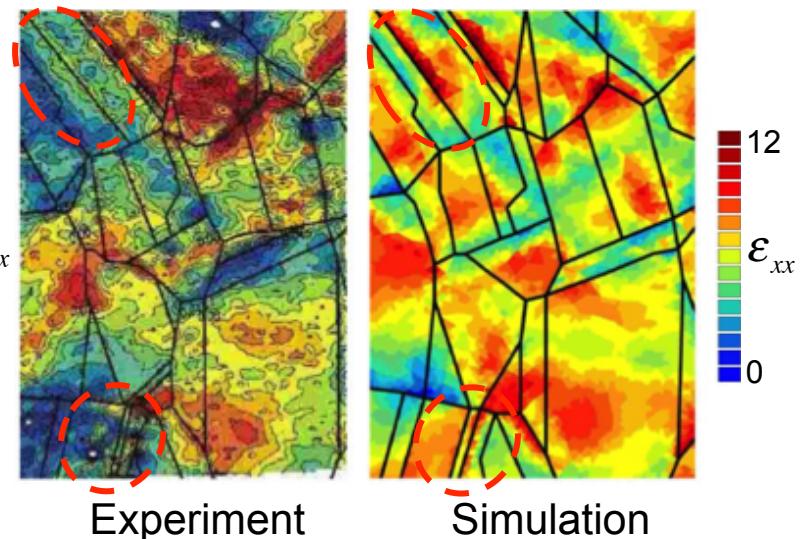
- **Introduction**
- **Strain Field Analysis: HR-DIC vs. CP-FEM**
- **Texture Analysis: EBSD vs. CP-FEM**
- **Summary**

“Comparisons between the model and experiments”

Zr 702 polycrystal (Heripre et al, 2007) (2.5 % strain)

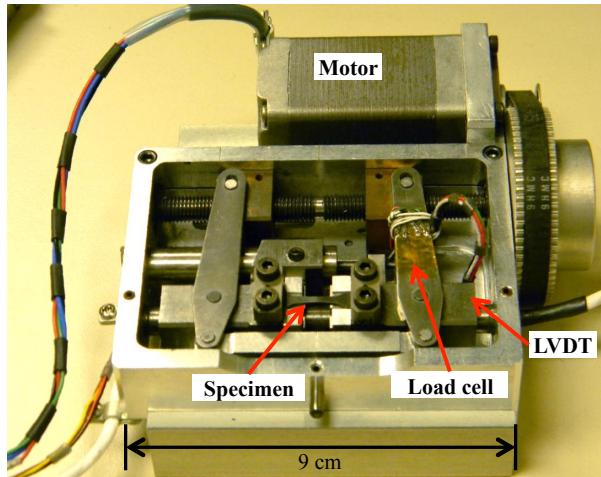


OFHC Cu polycrystal (Musienko et al, 2007) (5 % strain)

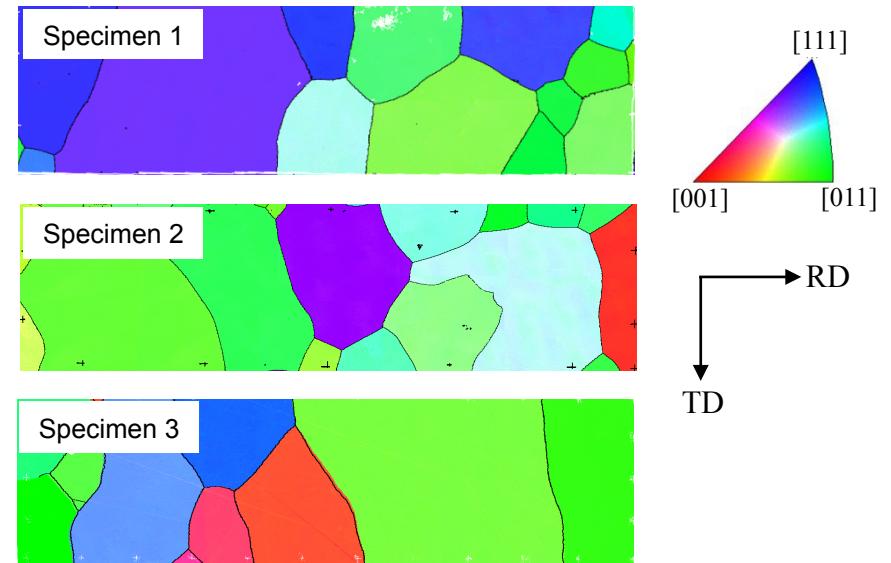
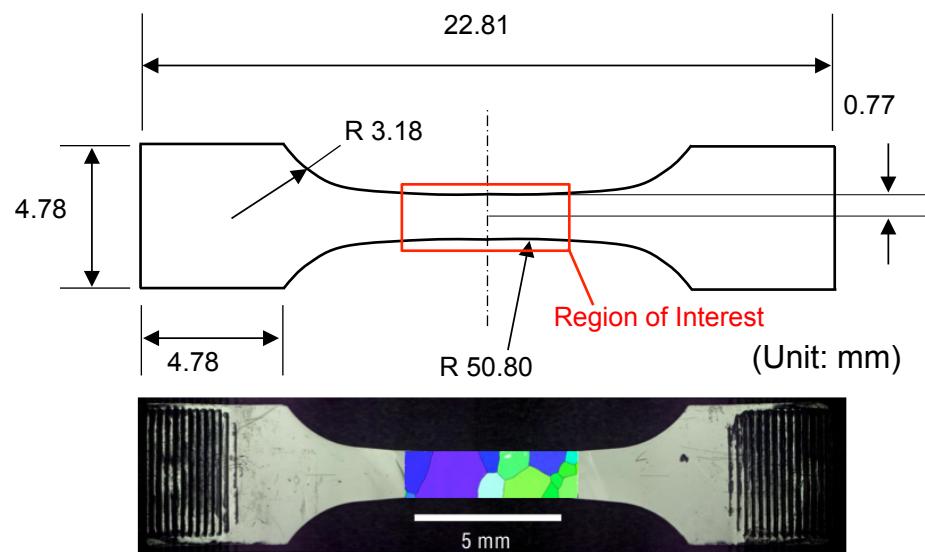


Limited success has been achieved in modeling polycrystal deformation behavior due to unknown subsurface grains

Experimental Setup

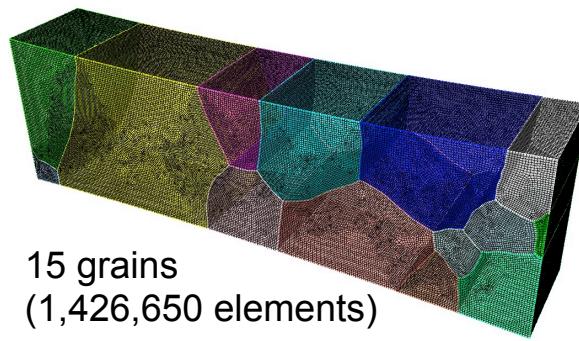


- Tantalum oligocrystals with mostly columnar 2D grain structure eliminate unknown subsurface grain morphology.
- *In-situ* load frame developed at Sandia
- HR-DIC (surface strain fields) and EBSD (crystal orientations) measurements at load inside SEM

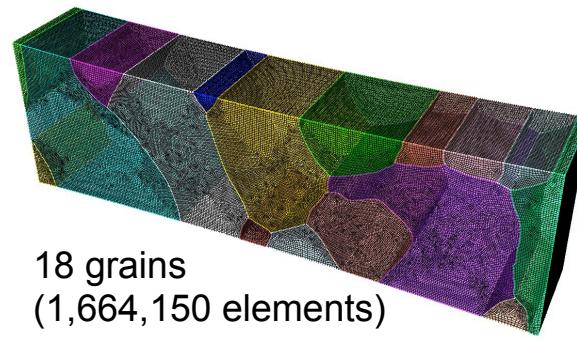


CP-FEM Simulations

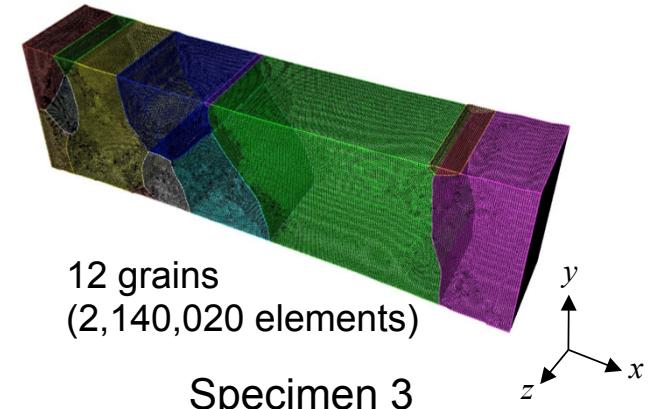
- BCC CP-FEM model developed at Sandia (JAS3D)*
- 24 $\{110\}<111>$ slip systems
- Rate dependent/ dislocation slip based: $\dot{\gamma}^\alpha = \dot{\gamma}_0^\alpha \left(\frac{\tau^\alpha}{g^\alpha} \right)^{1/m}$
- Slip resistance: $g^\alpha = \min(\tau_{EI}^{*\alpha}, \tau_{LT}^{*\alpha}) + \tau_{obs}^\alpha$
- Strain hardening: $\tau_{obs}^\alpha = A\mu b \sqrt{\sum_{\beta=1}^{NS} \rho^\beta} \quad \dot{\rho}^\alpha = \left(\kappa_1 \sqrt{\sum_{\beta=1}^{NS} \rho^\beta} - \kappa_2 \rho^\alpha \right) \cdot |\dot{\gamma}^\alpha|$



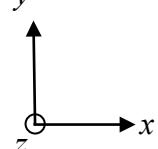
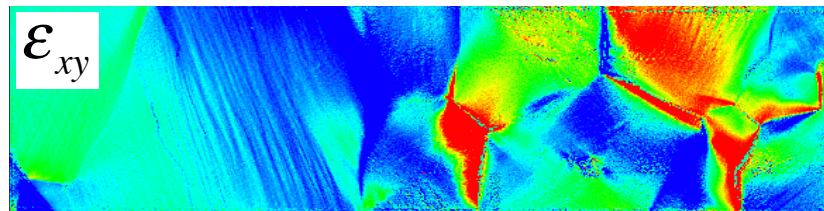
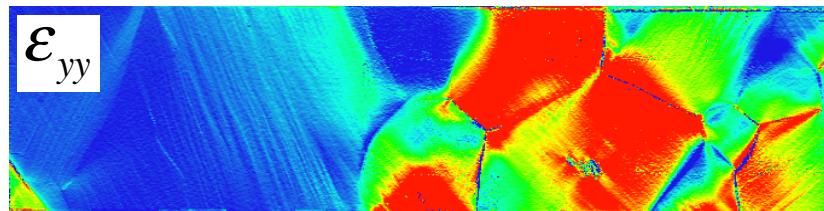
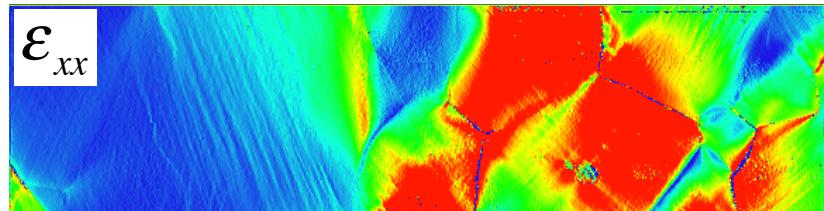
Specimen 1



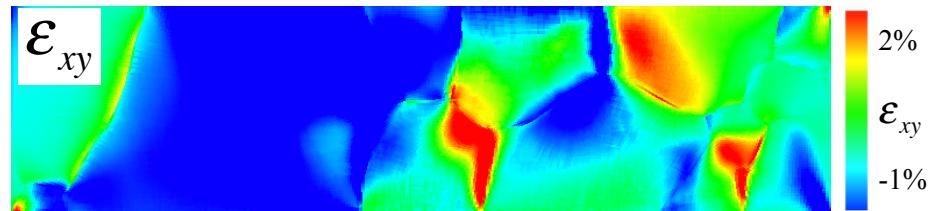
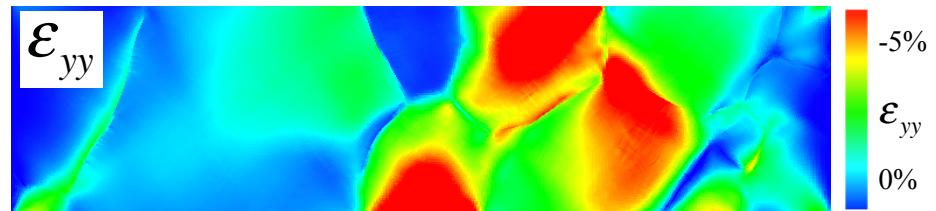
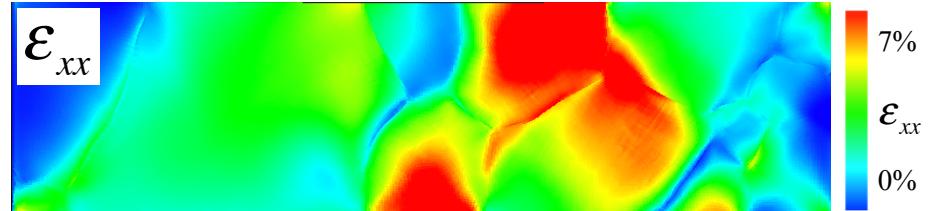
Specimen 2



HR-DIC measurements



CP-FEM predictions

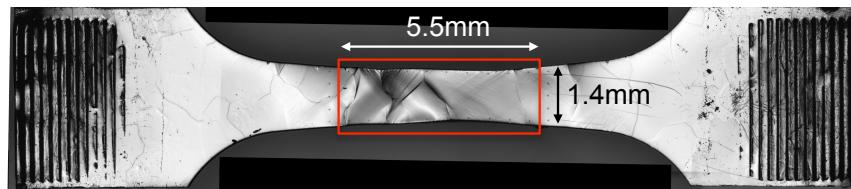


Measured and predicted strain fields agree reasonably well

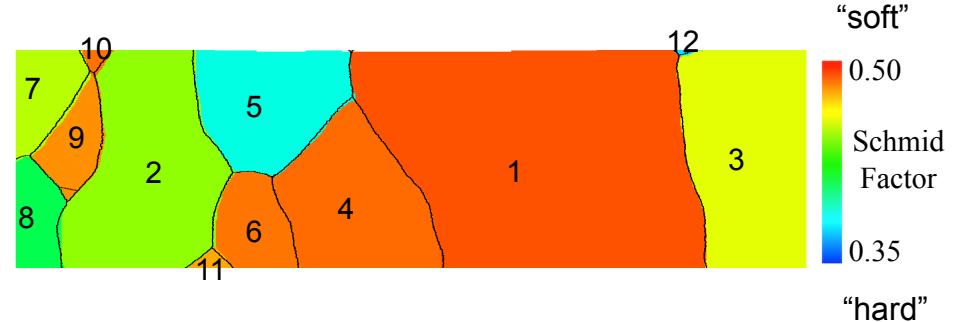
Applied strain = 4.3%

Surface Strain Field Predictions: Specimen 3

Deformed specimen



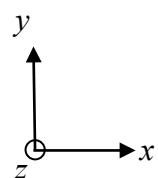
Max. Schmid factor



Measured strain fields (HR-DIC)



Predicted strain fields (CP-FEM)



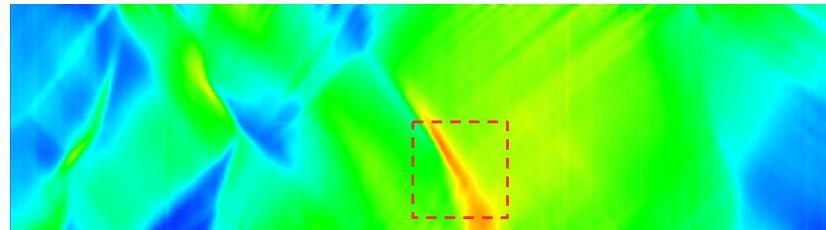
Applied strain = 0 %

Measured and predicted strain fields agree reasonably well

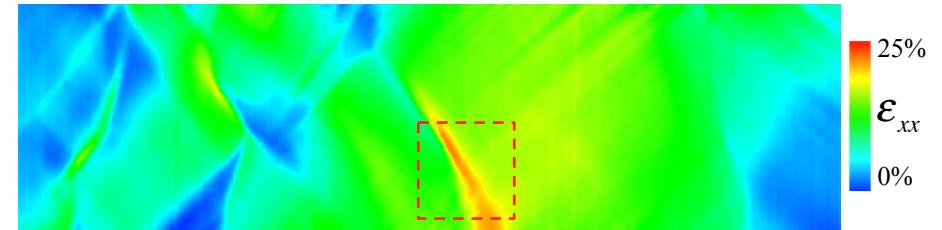
Rescaling of DIC Data: Specimen 3

Projection of raw HR-DIC data onto the finite element mesh

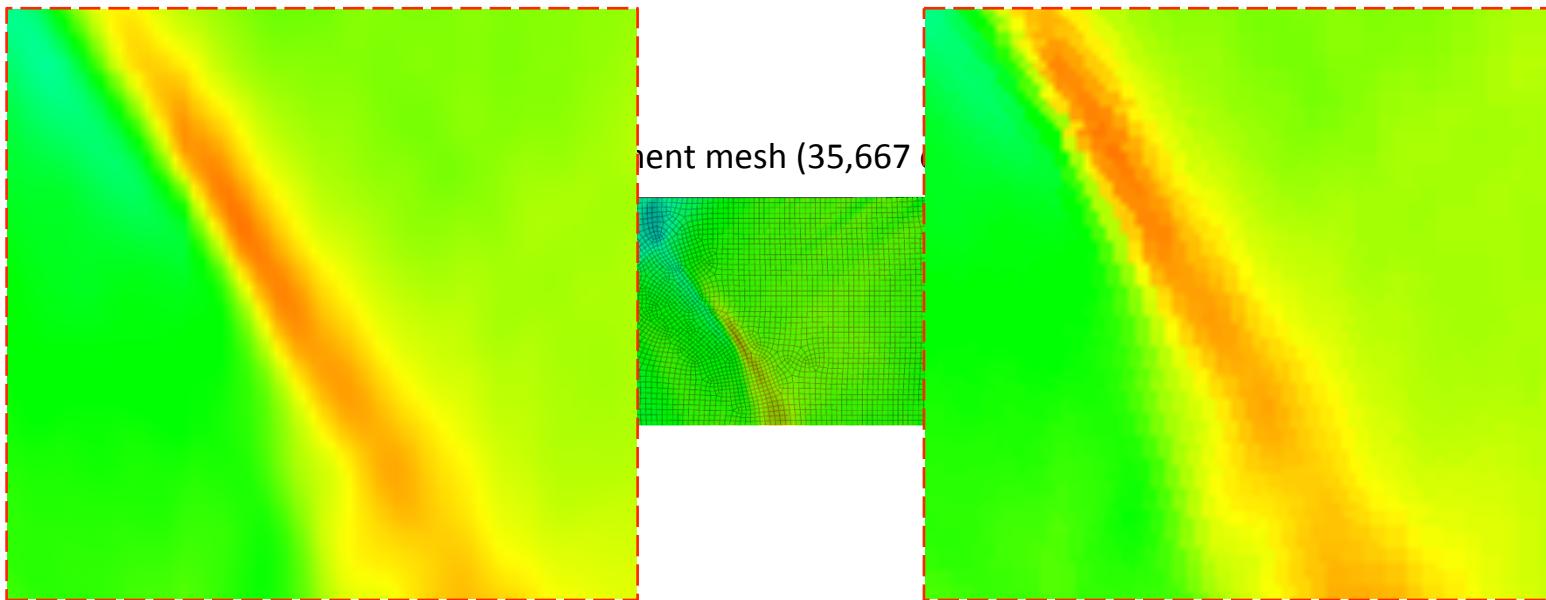
Full DIC data (781,015 data points)



Reduced DIC data (35,667 data points)



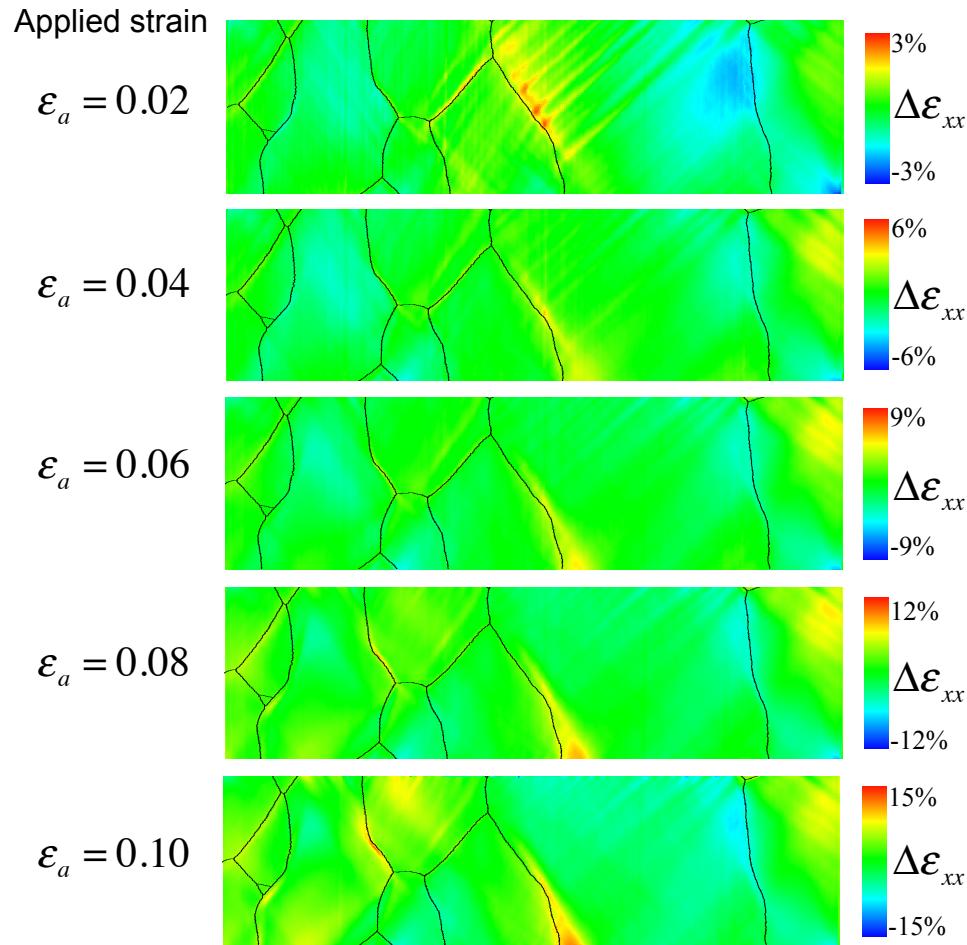
Finite element mesh (35,667



Quantitative Sim.-Exp. Comparisons

Deviation between measured and simulated ε_{xx}

$$\Delta\varepsilon = \varepsilon^{DIC} - \varepsilon^{sim} \quad (\text{per element basis})$$

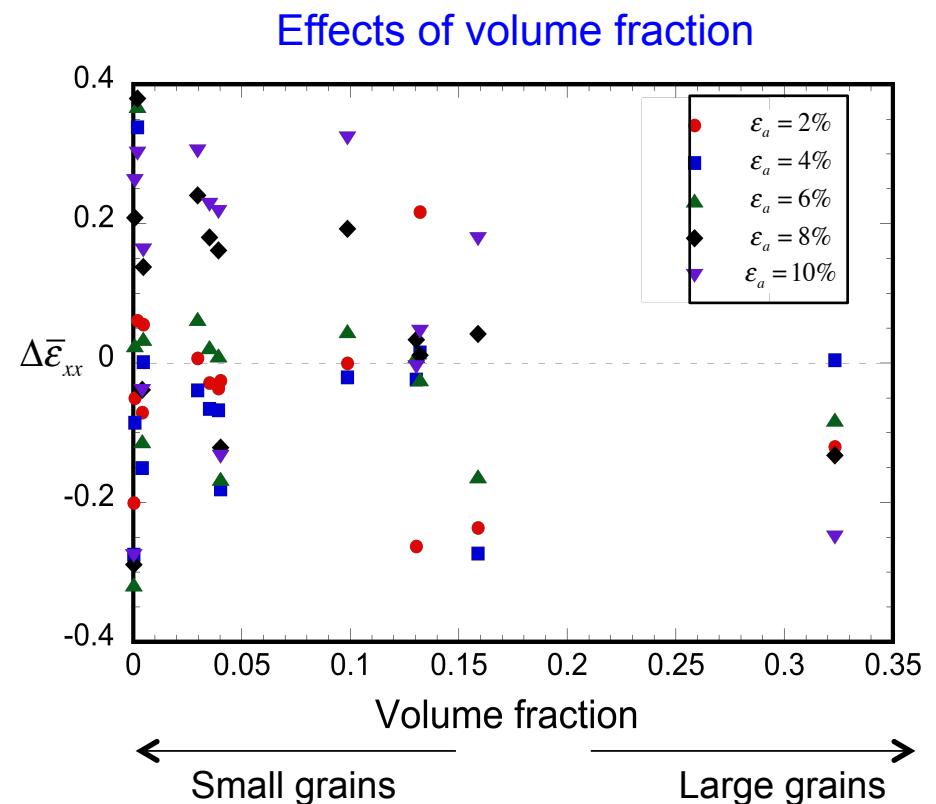
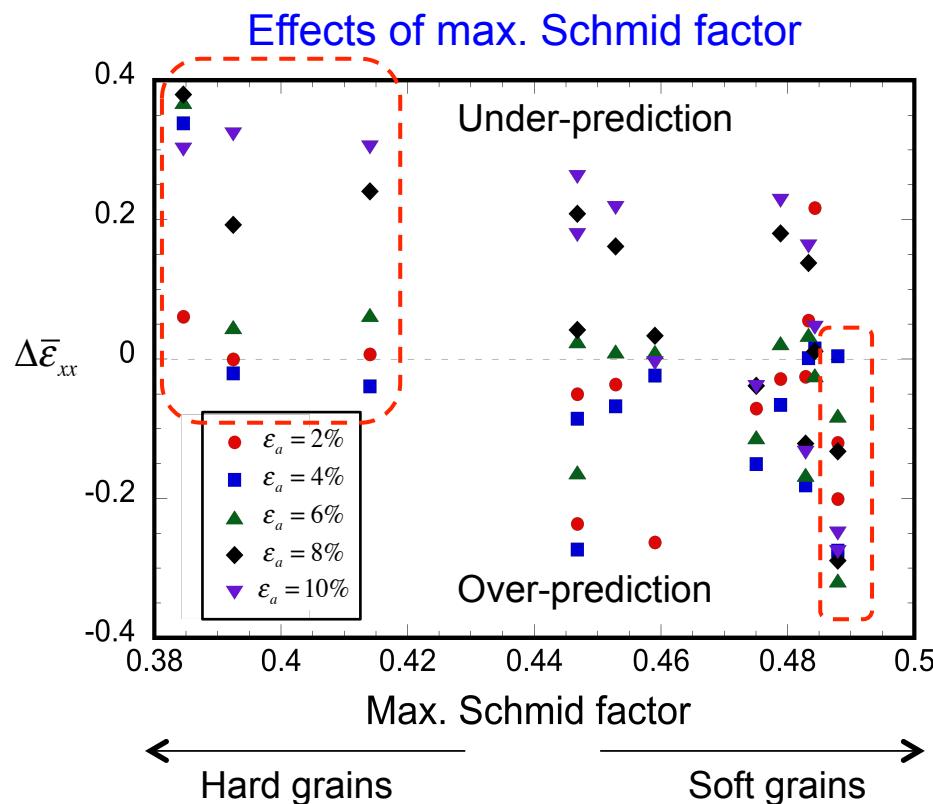


$$\Delta\varepsilon^{avg} = \sqrt{\frac{1}{N} \sum_{i=1}^N (\varepsilon_i^{DIC} - \varepsilon_i^{sim})^2}$$

Applied strain	$\Delta\varepsilon_{xx}^{avg}$	$\Delta\varepsilon_{yy}^{avg}$	$\Delta\varepsilon_{xy}^{avg}$
0.02	0.007	0.006	0.003
0.04	0.010	0.010	0.007
0.06	0.014	0.012	0.010
0.08	0.021	0.014	0.014
0.10	0.034	0.018	0.016

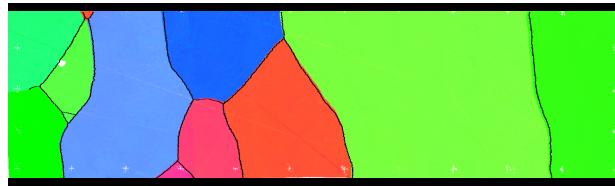
Effects of Crystal Orientations and Grain Sizes

$$\Delta \bar{\varepsilon}_{xx} = \frac{\varepsilon_{xx}^{DIC} - \varepsilon_{xx}^{sim}}{\varepsilon_a} \text{ (per grain basis)}$$

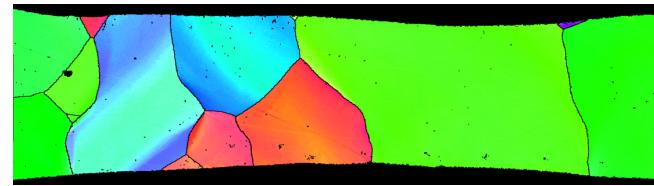


CP-FEM predictions are less successful for grains in extreme orientations

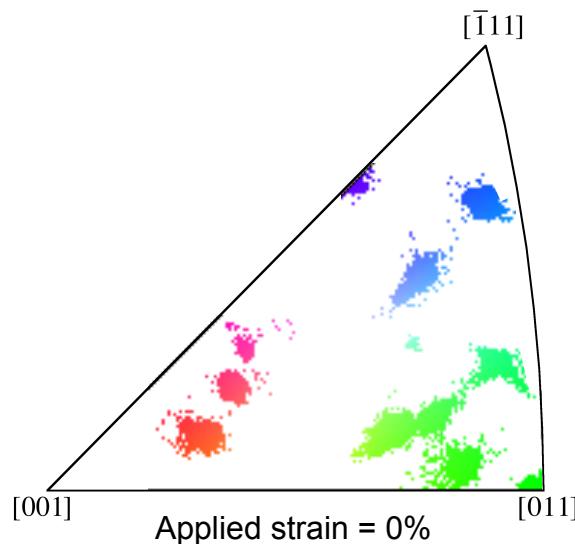
Crystal Rotations: IPFs in RD



Applied strain = 0%

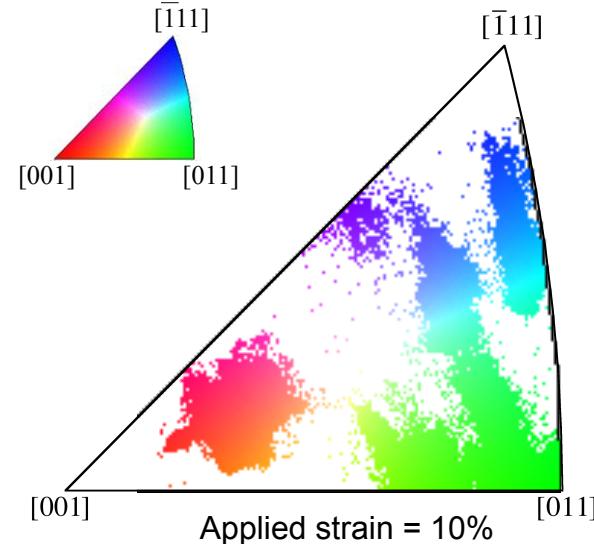


Applied strain = 10%

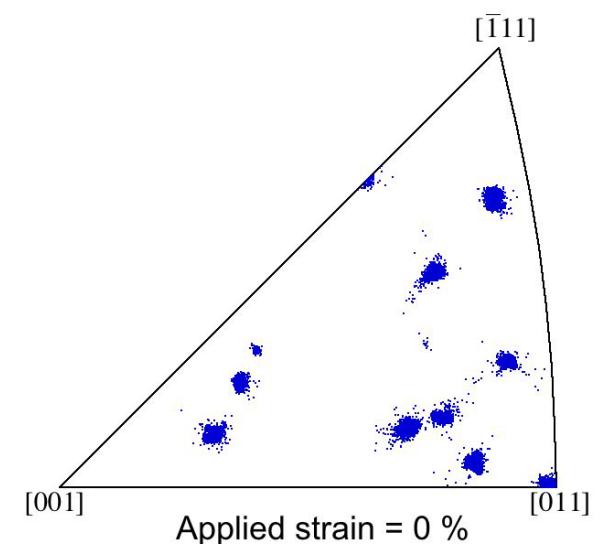


Applied strain = 0 %

EBSD measurement



Applied strain = 10%

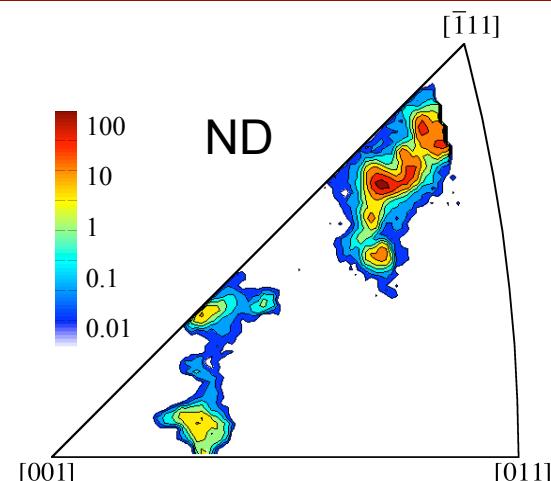
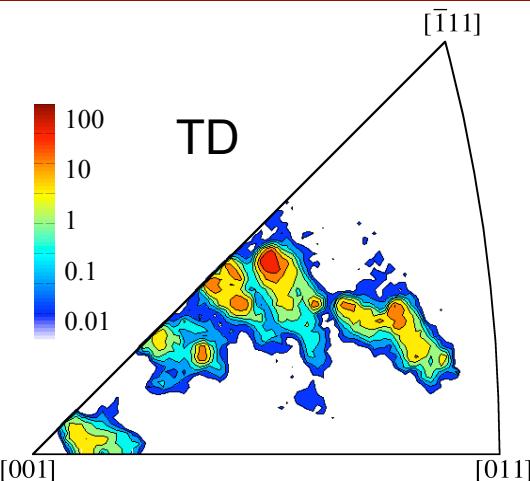
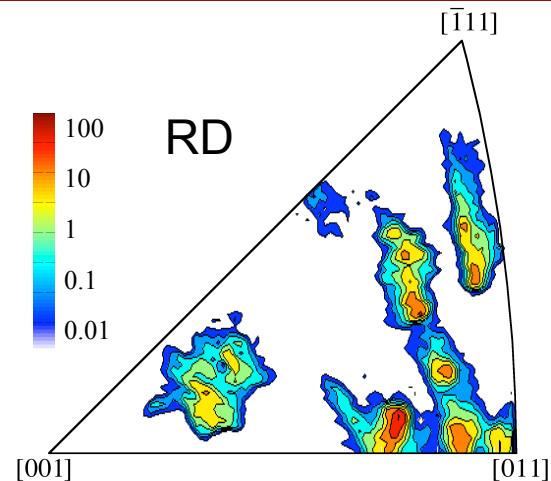


Applied strain = 0 %

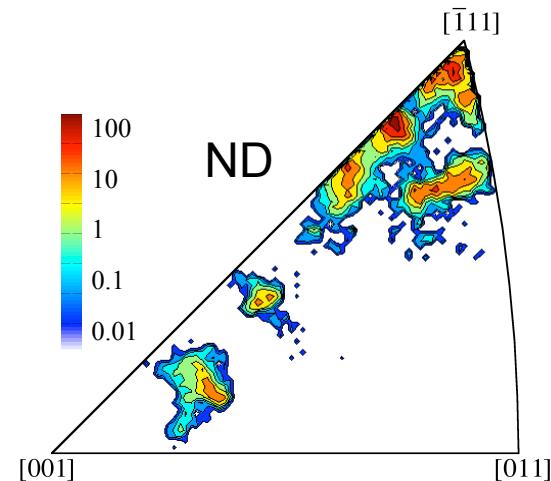
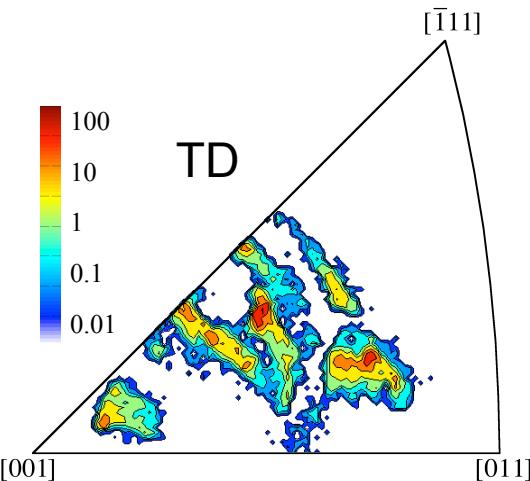
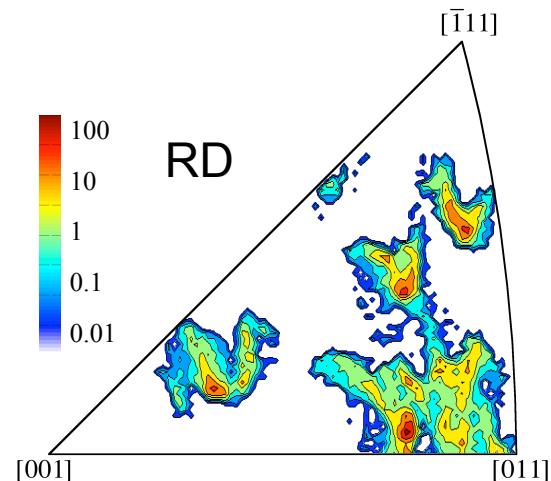
CP-FEM prediction

IPF contour plots indicate good agreement between model and experiment.

Deformed Texture Predictions



EBSD measurements



CP-FEM predictions

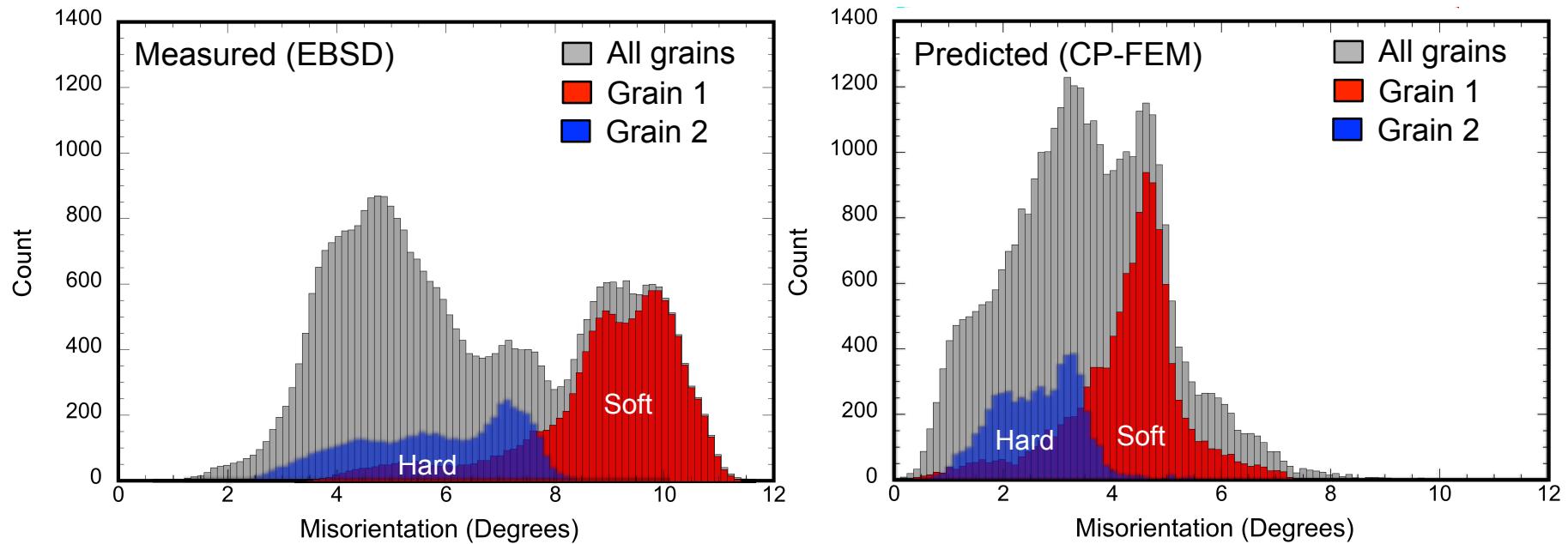
IPF contour plots indicate good agreement between model and experiment.

Crystal Rotations: Misorientation

$$\theta = \cos^{-1} \left[\frac{1}{2} (\Delta g_{11}^{AB} + \Delta g_{22}^{AB} + \Delta g_{33}^{AB} - 1) \right] \quad \Delta g^{AB} = g^B (g^A)^{-1} \quad g : \text{rotation matrix}$$



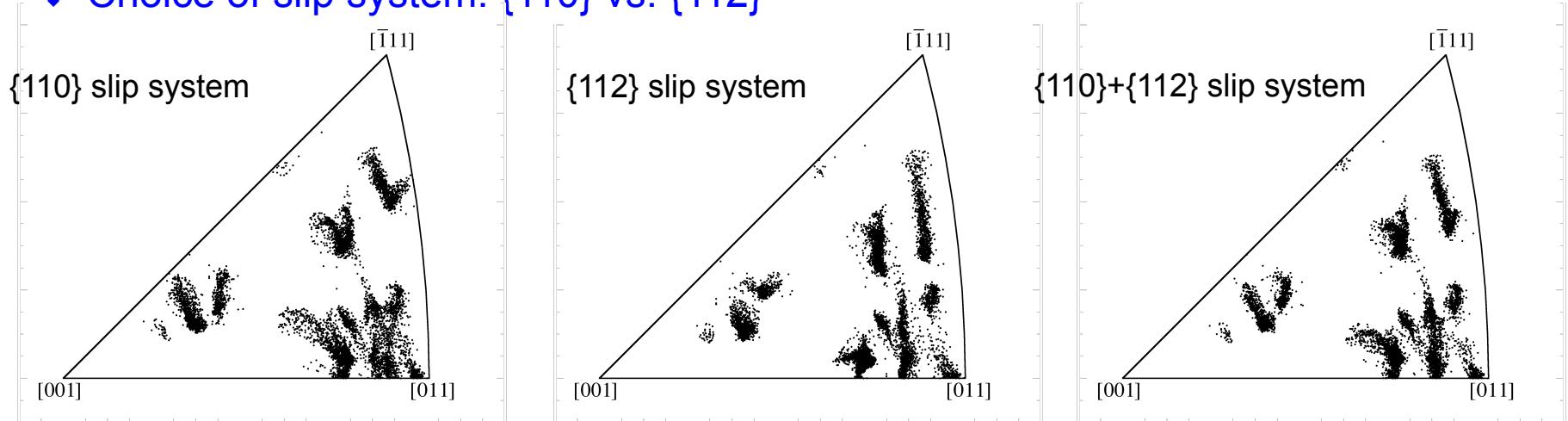
Misorientation angles (at 10% deformation) relative to the initial orientation



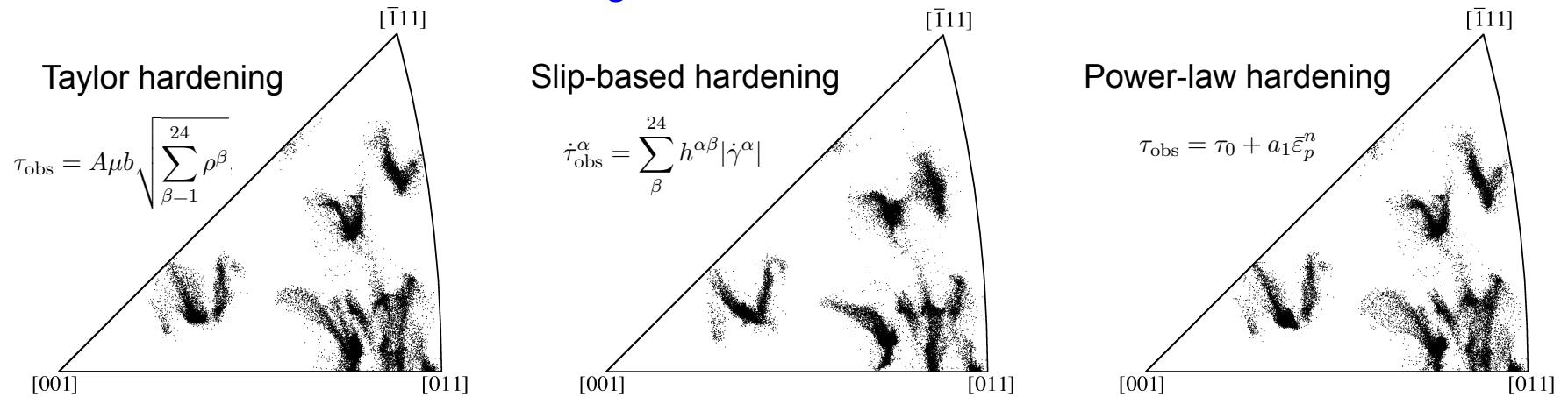
CP-FEM model under-predicted measured crystal rotation

Other Constitutive Effects

- ❖ Choice of slip system: $\{110\}$ vs. $\{112\}$

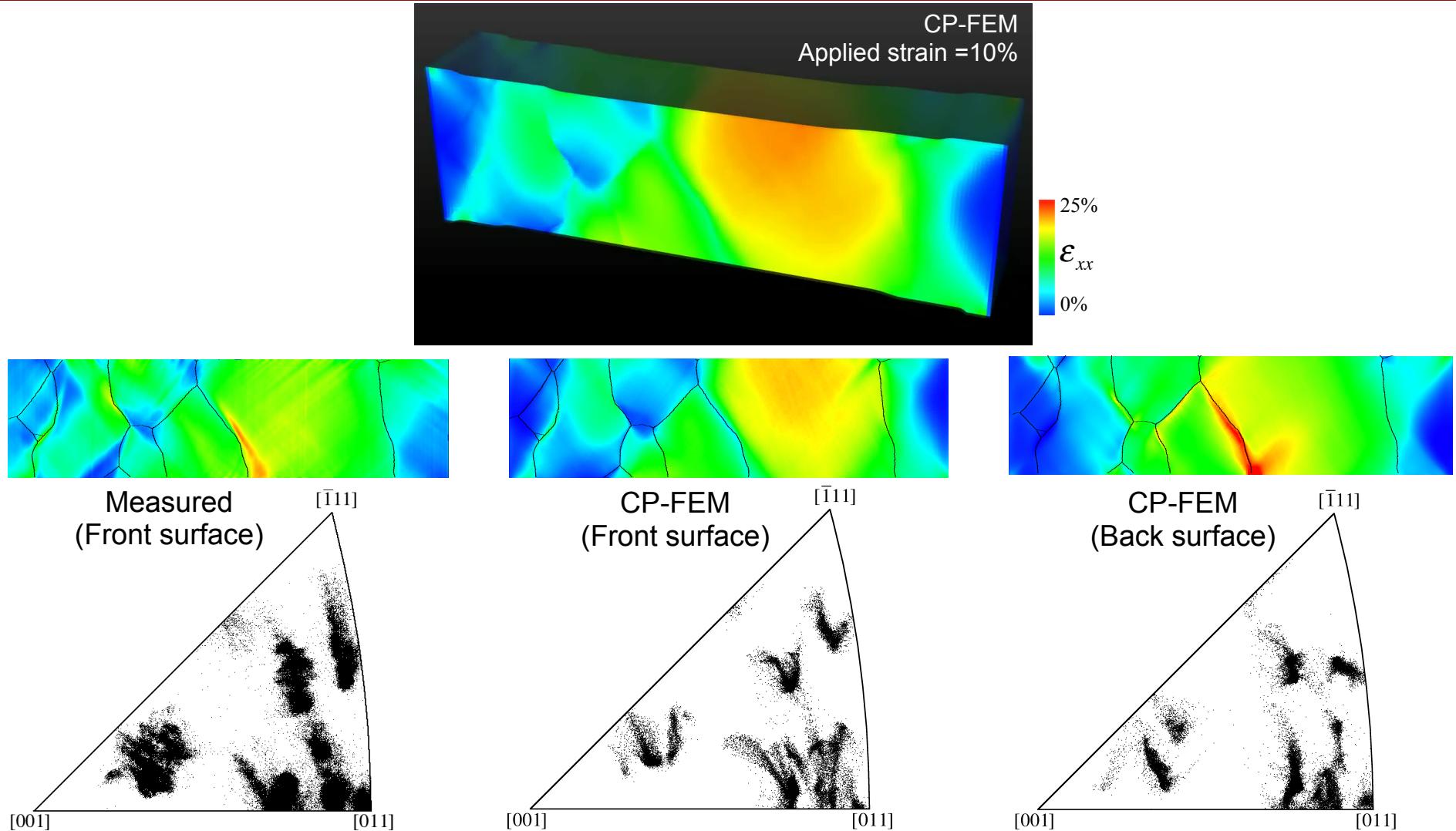


- ❖ Effects of different hardening laws



Choice of slip system and hardening law has relatively small effect

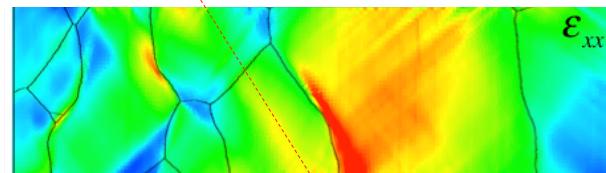
3D Effects



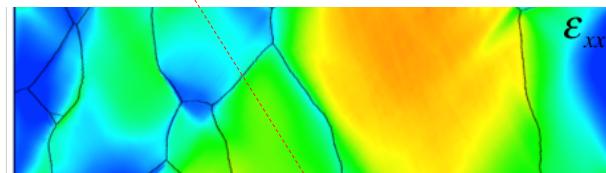
Full 3D model is required to accurately model columnar structured specimens.

Limitations: Fracture

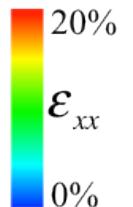
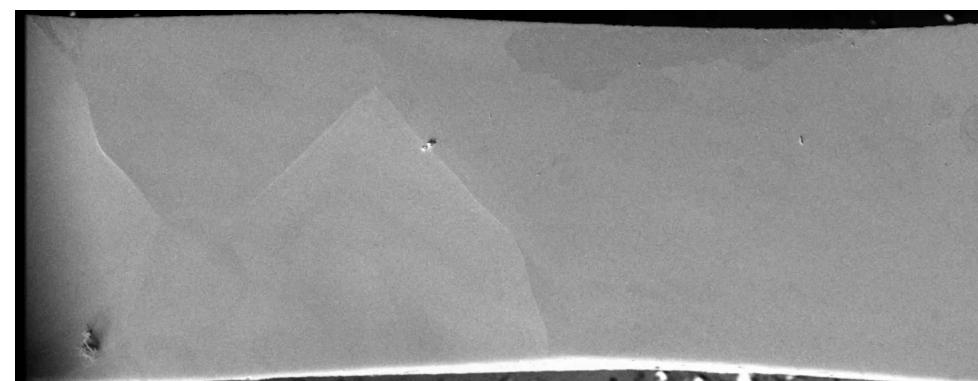
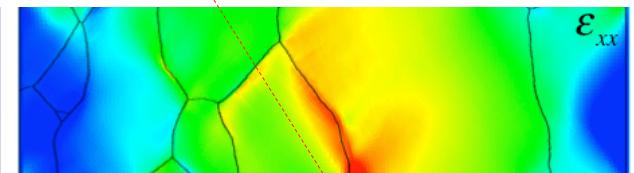
Experiment (front)



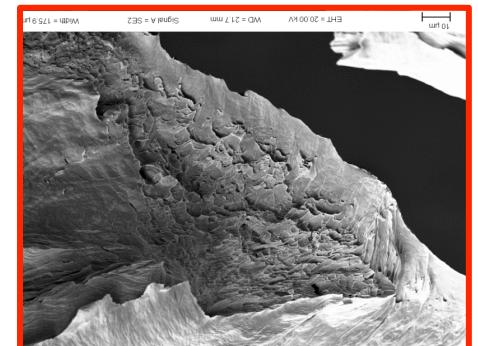
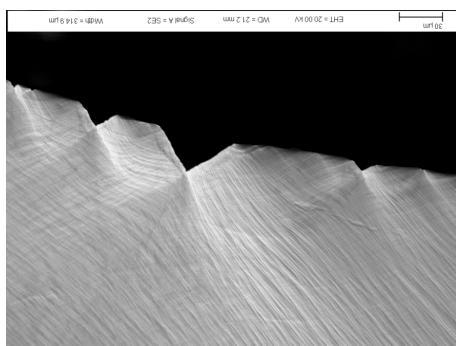
Model (front)



Model (back)



Selected images from various points in the fracture process.



Summary

Conclusions

- First quantitative comparison between CP-FEM simulations and experimental measurements
- Model predictions of strain fields showed good agreement with HR-DIC measurements (~ 3% deviation at 10% applied strain).
- Model predictions of crystal rotations showed good agreement with EBSD measurements. Larger deviations at extreme crystal orientations.

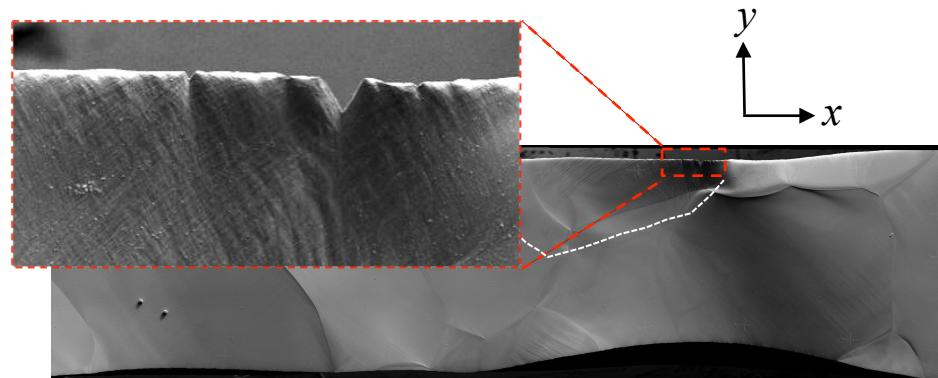
Future Work

- More sophisticated treatment of grain boundary and dislocation – grain boundary interactions
- Damage/ Fracture modeling

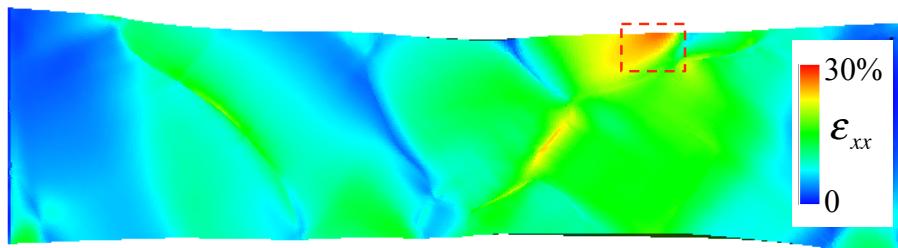
THANK YOU !

Email: hnlim@sandia.gov

Failure Predictions: Specimen 2

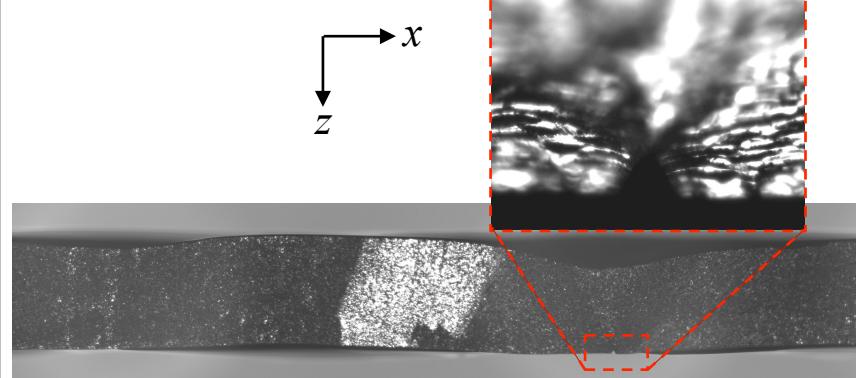


Surface image (19.2%)

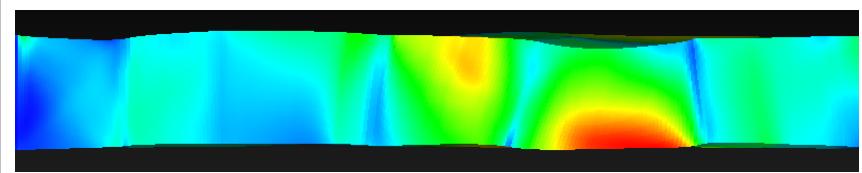


Simulated ϵ_{xx}

Side view



Surface image (19.2%)



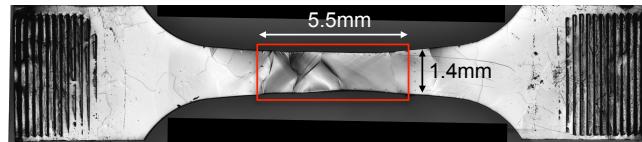
Simulated ϵ_{xx}

Top view

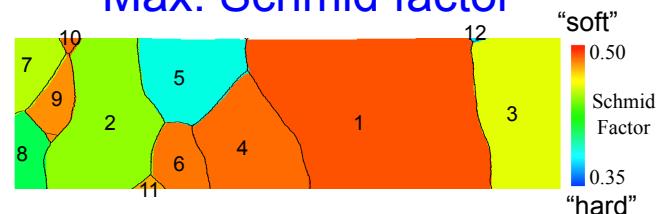
Failure location agrees with the location of the highest ϵ_{xx} from the simulation

Surface Strain Field Predictions: Specimen 3

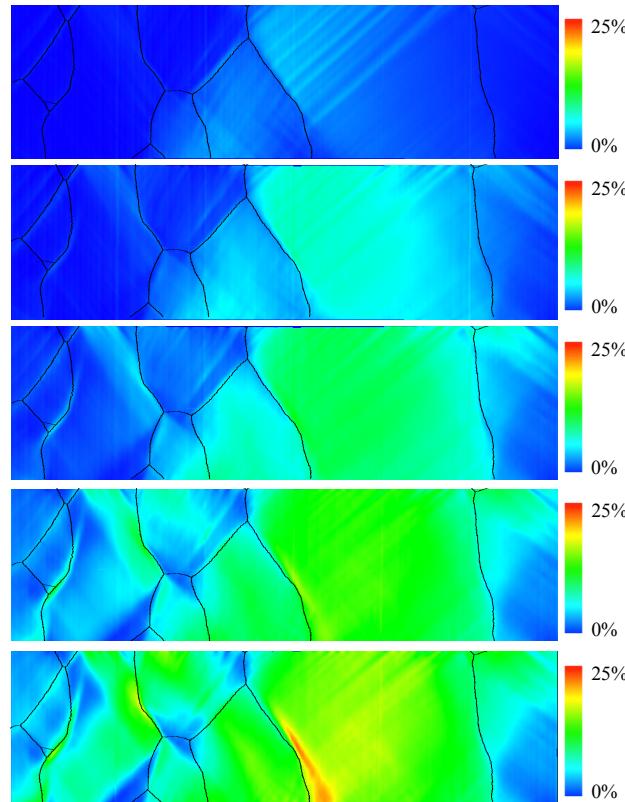
Deformed specimen



Max. Schmid factor

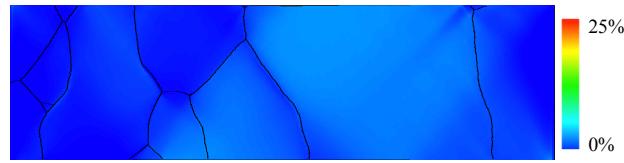


Measured strain fields (HR-DIC)

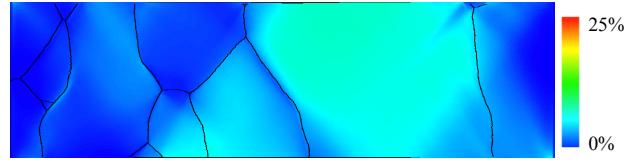


Predicted strain fields (CP-FEM)

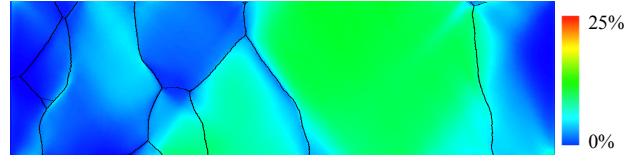
$$\varepsilon_a = 2\%$$



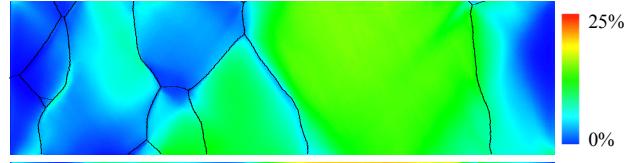
$$\varepsilon_a = 4\%$$



$$\varepsilon_a = 6\%$$



$$\varepsilon_a = 8\%$$



$$\varepsilon_a = 10\%$$

