

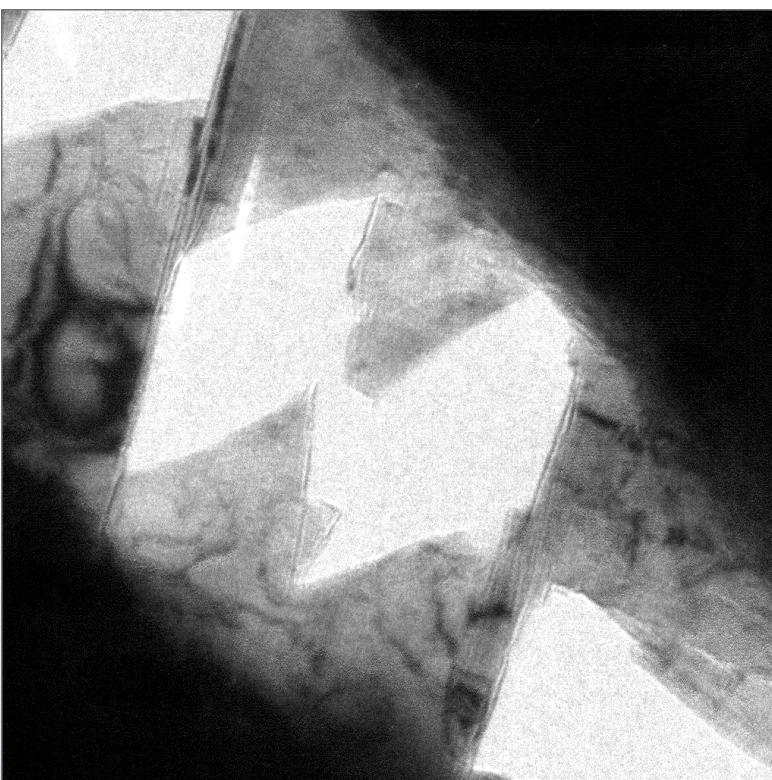
Comparison of Hydrogen Introduction Techniques for In-situ TEM Straining Mechanisms

SAND2018-2667C

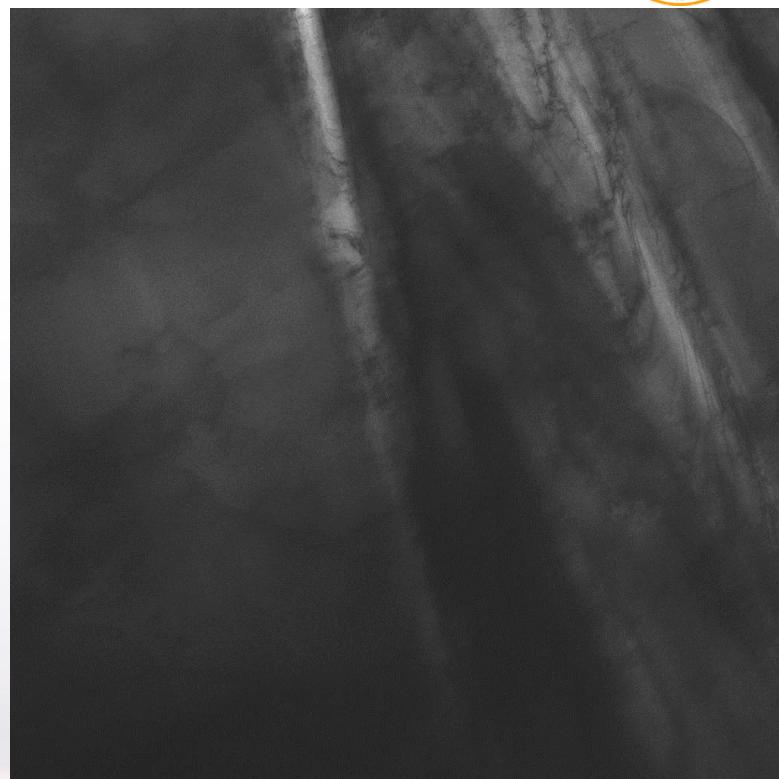
Khalid Hattar, Christopher M. Barr, Daniel Bufford, Brittany Muntifering,

Kathryn A. Small, Ai Leen Koh, Rick Karneskey

Sandia National Laboratories & Stanford University



March 13, 2018



Exploring the various methods to study hydrogen embrittlement in the TEM



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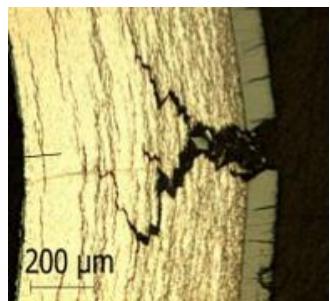
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Fracture is a Costly Problem Outside of Sandia That Remains Unsolved

R81

FRACTURE COSTS US \$119 BILLION A YEAR, SAYS STUDY BY BATTELLE/NBS*

J.J. Duga, W.H. Fisher, R.W. Buxbaum, A.R. Rosenfield, A.R. Buhr, E.J. Honton, and S.C. McMillan



Even if the probability of cladding failure from this cause during the lifetime of a fuel element is low ($\sim 10^{-5}$), the consequences are not ($\sim \$1\text{M/day}$ if shutdown of the reactor is required and replacement power must be purchased) [Olander, JNM, 2009].

Hydrogen tank leak evacuates area near Pixar in Emeryville

Bay Bridge bolt fix could cost \$10 million, opening date still unclear

By Isabel Angell



Emeryville hydrogen tank leak

The Seattle Times

Winner of Eight Pulitzer Prizes

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Originally published October 4, 2010 at 11:07 AM | Page modified October 5, 2010 at 11:25 AM

Comments (0) E-mail article Print Share

State declares Tesoro blast was preventable

The Washington state Department of Labor & Industries on Monday fined Tesoro a record \$2.39 million after a deadly explosion in April at the oil refinery in Anacortes.

By Susan Gilmore and Craig Welch
Seattle Times staff reporters

MOUNT VERNON — As the state Monday assessed a record \$2.39 million fine against Tesoro for last

◀ PREV 1 of 2 NEXT ▶

Hydrogen Embrittlement of Metals

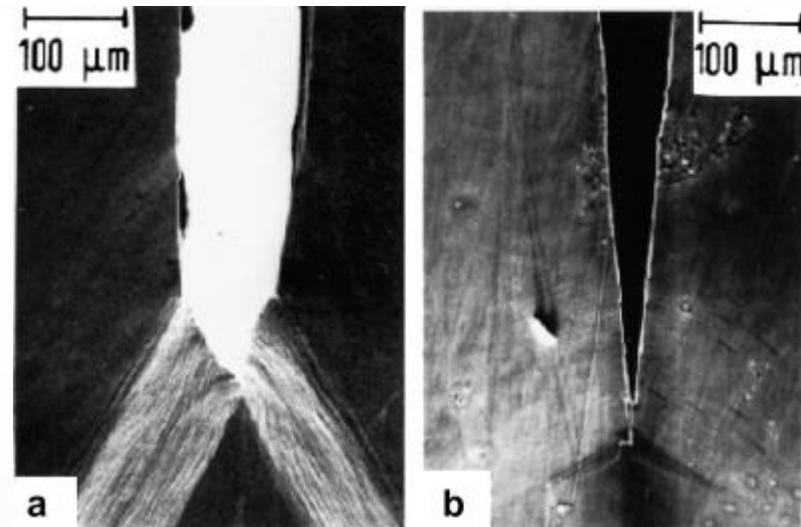
Collaborator: for work on this slide

Hydrogen content degrades mechanical properties of metal systems

- Loss in ductility, strength, and toughness
- Proposed Mechanisms
 - Hydrogen-enhanced decohesion (HEDE)
 - Hydrogen-enhanced localized plasticity (HELP)
 - Hydride formation

Accumulation at microstructural interfaces

- H induced cracks seen in steels, nickel and nickel based alloys
 - Doesn't require stress or elevated temperature
- H-dislocation interaction observed to be prelude to embrittltment



Crack tip opening angle in Ni single crystal decreases as H pressure increases

Louthan Jr., M.R. Savannah River National Laboratory

Barnoush, A. et al. (2008). *Corrosion Science*.

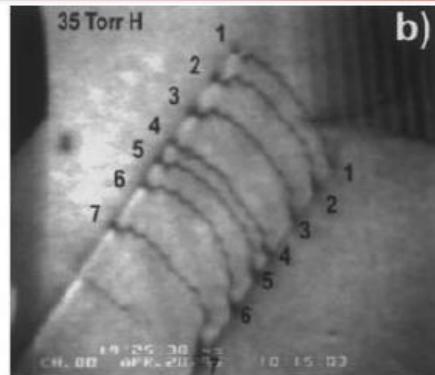
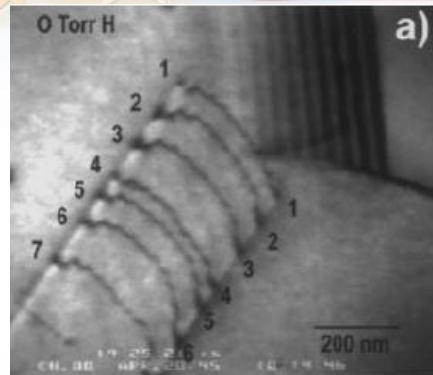
Matusiewicz, G. (1985). *Acta Metallurgica*.

In-situ TEM techniques provide insight to mechanism of failure due to H embrittlement



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20 years of *In situ* TEM Hydrogen Studies

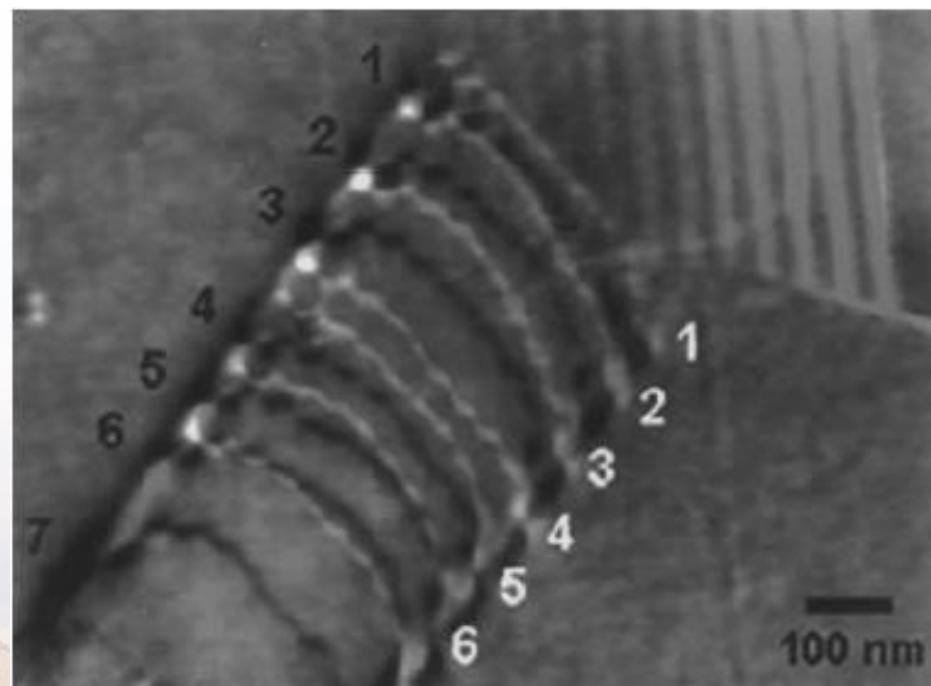
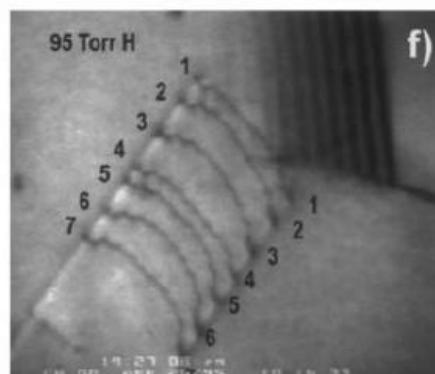
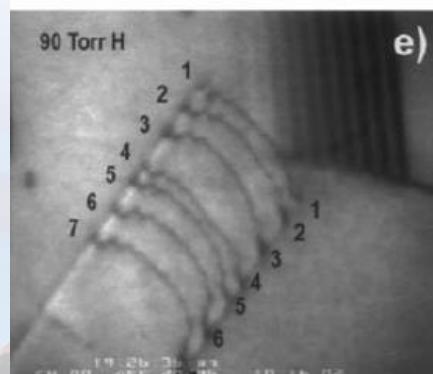
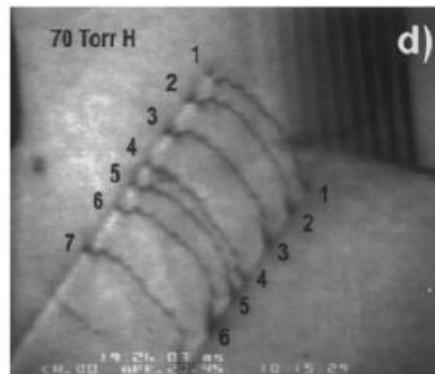
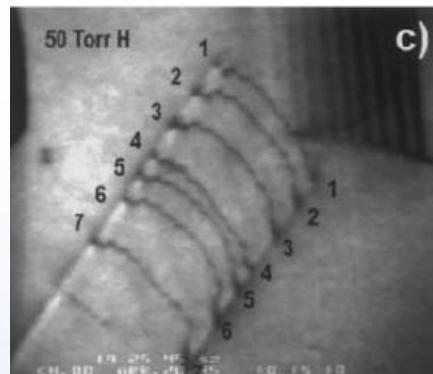


HYDROGEN EFFECTS ON THE INTERACTION BETWEEN DISLOCATIONS

P. J. FERREIRA[†], I. M. ROBERTSON and H. K. BIRNBAUM

University of Illinois, Department of Materials Science and Engineering, and Frederick Seitz Materials Research Laboratory, Urbana, IL 61801, U.S.A.

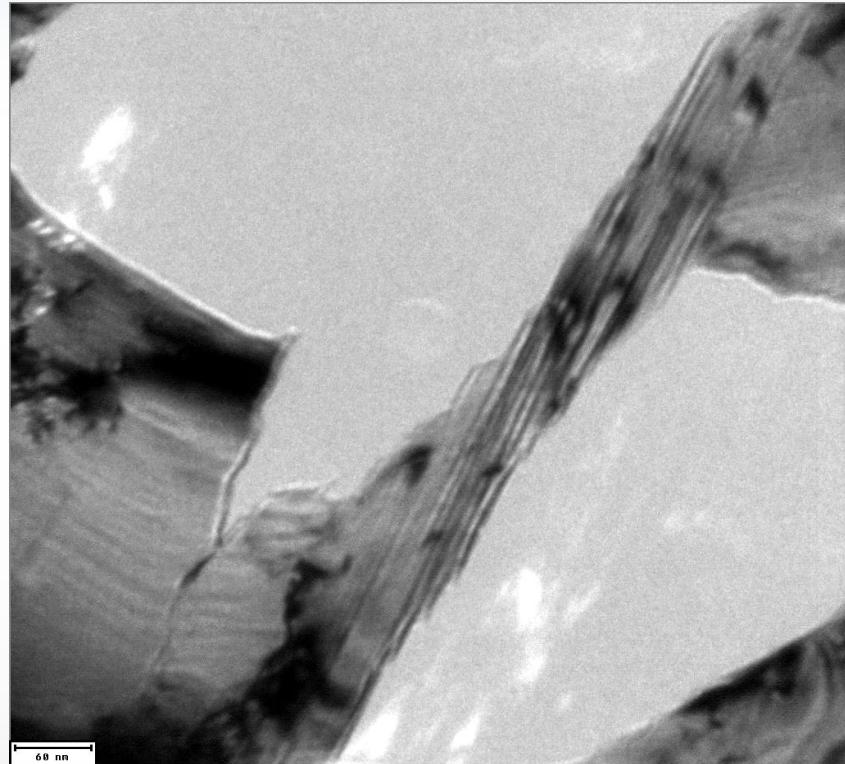
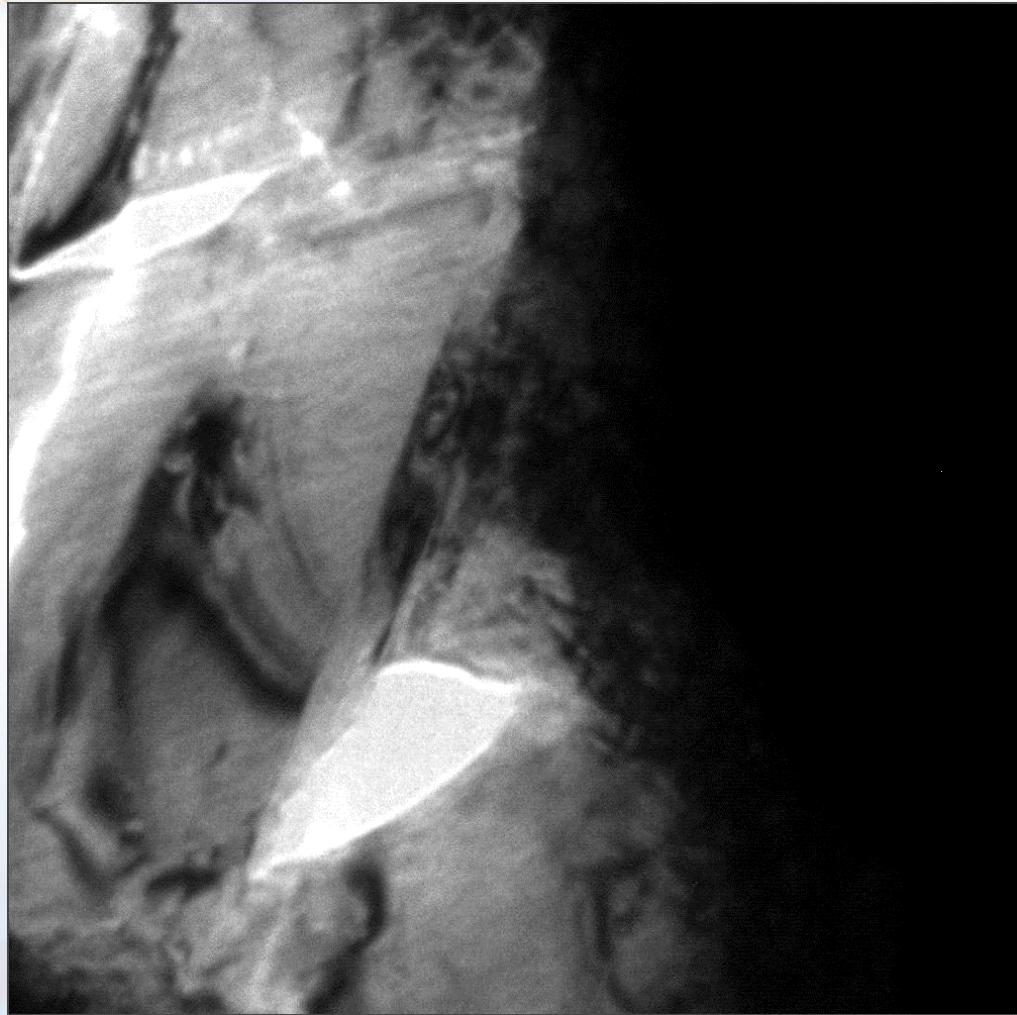
(Received 29 August 1997)



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Control Experiments: Failure of pure Ni

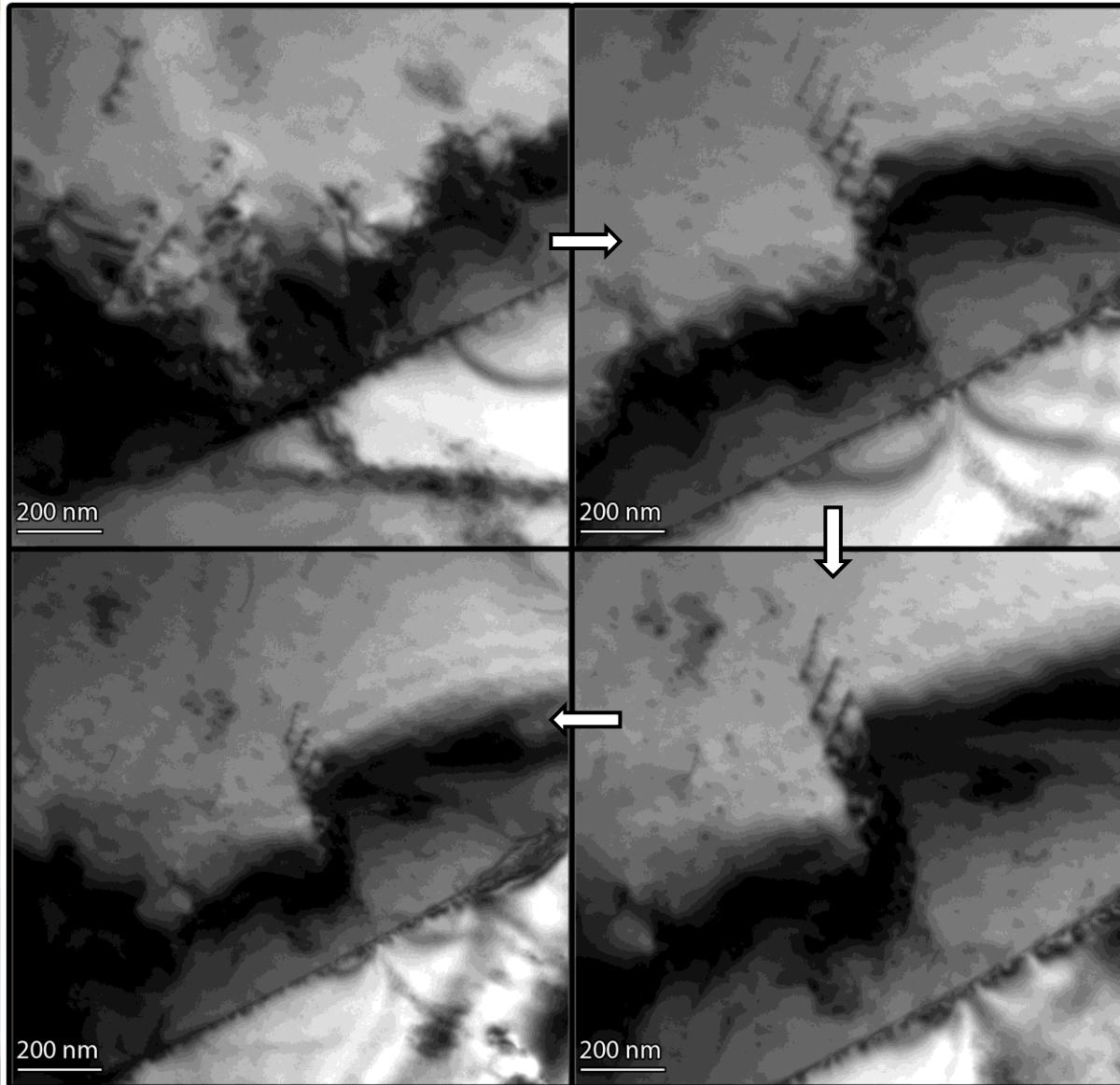


In-situ TEM techniques provide insight to mechanism of failure due to H embrittlement



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Traditional Charging and In-situ Straining



Dislocations appear to move toward and be absorbed into boundary

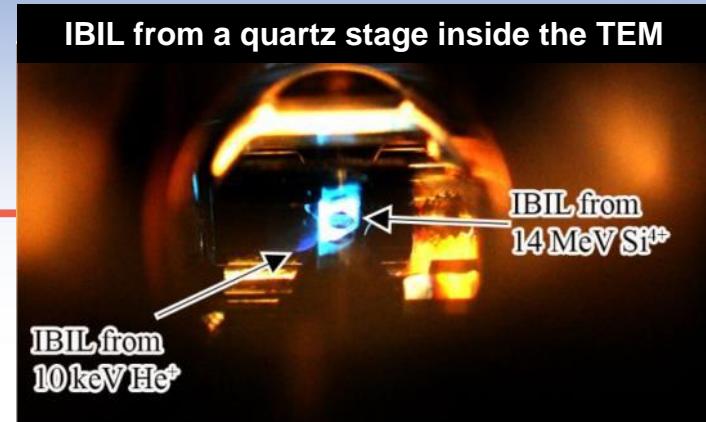
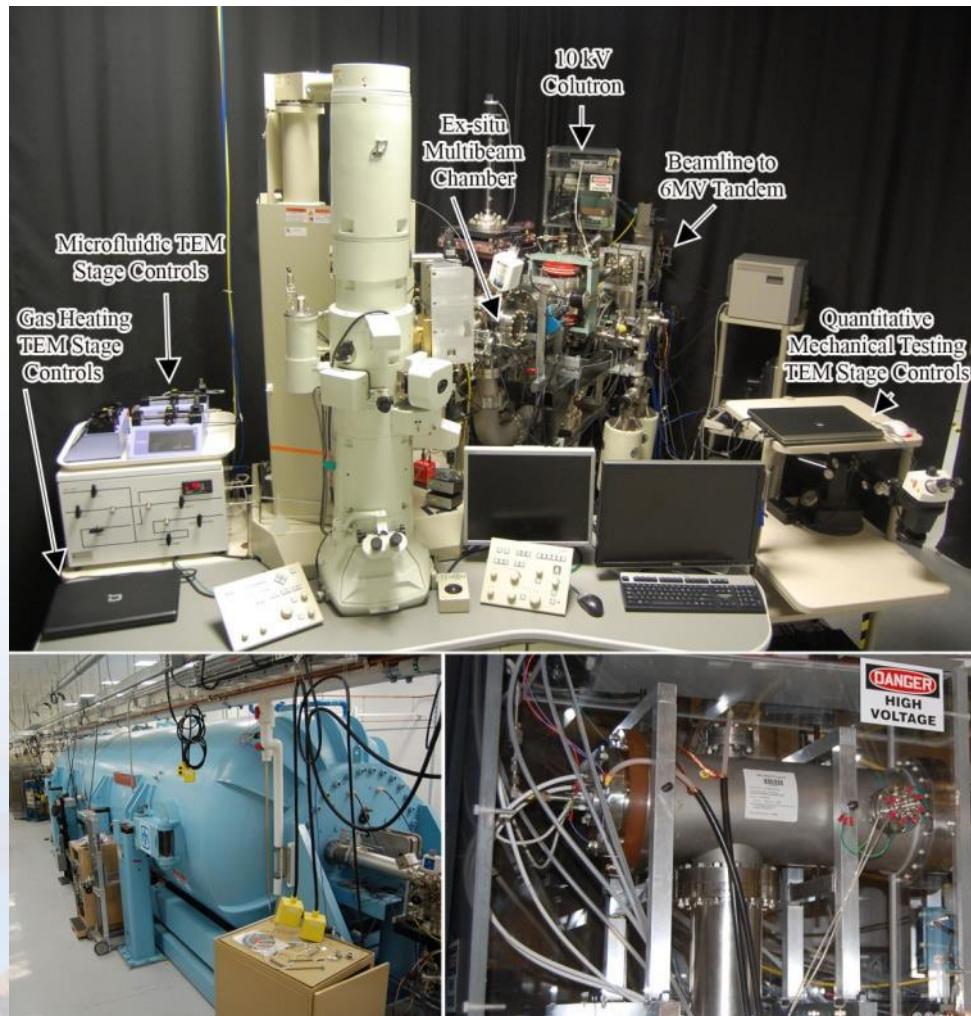


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Sandia's Concurrent *In situ* Ion Irradiation TEM Facility

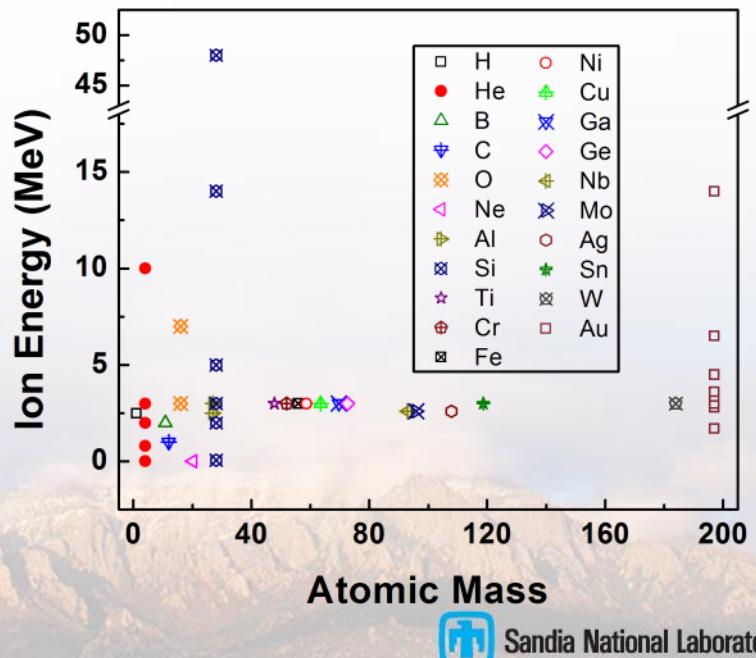
Collaborator: D.L. Buller

10 kV Colutron - 200 kV TEM - 6 MV Tandem



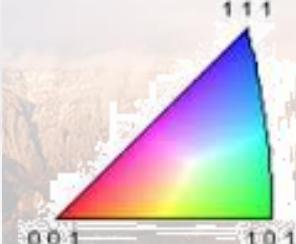
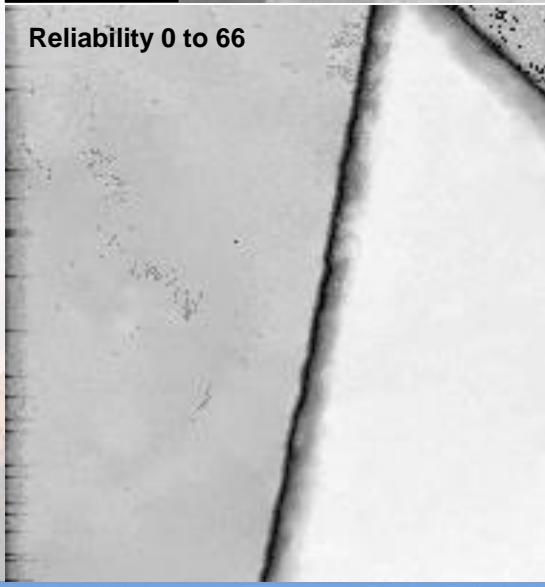
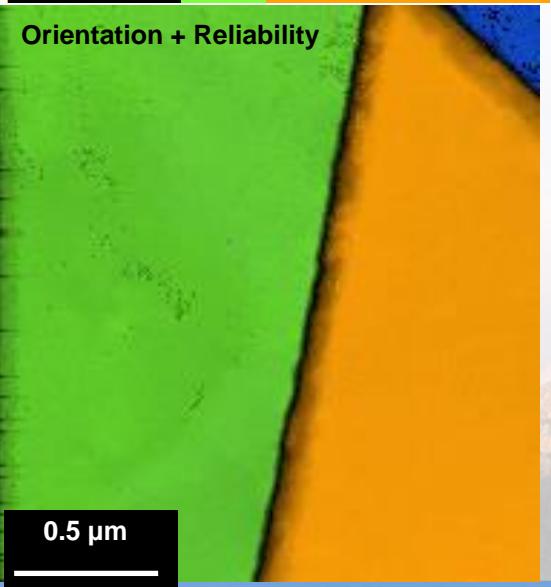
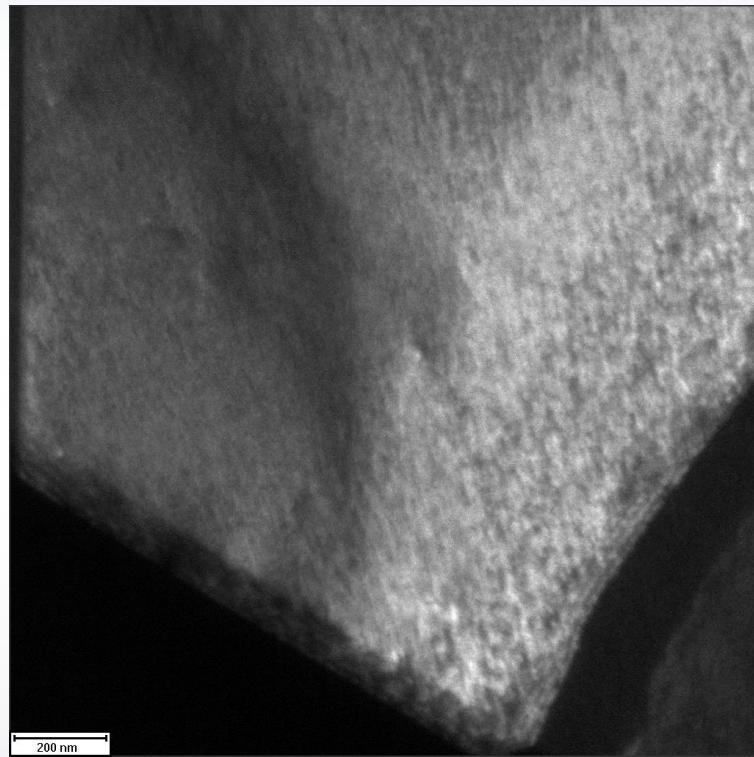
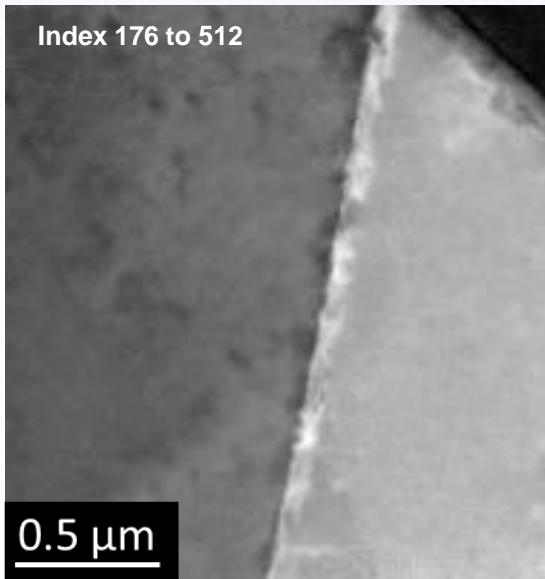
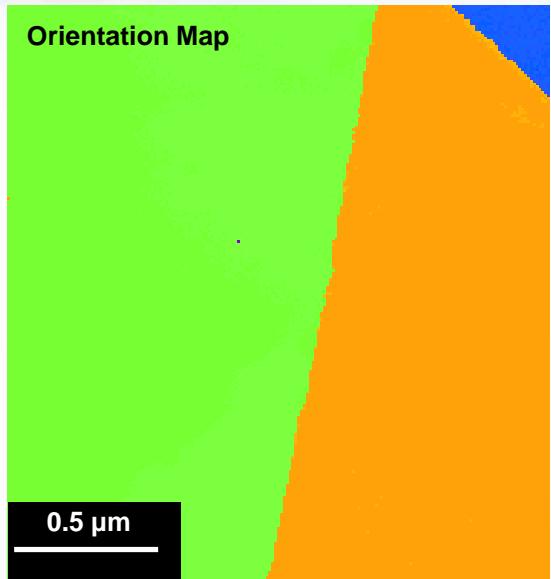
Direct real time observation
of ion irradiation,
ion implantation, or both
with nanometer resolution

Ion species & energy introduced into the TEM



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Characterization of Orientation, GB structure and Displacement Defects

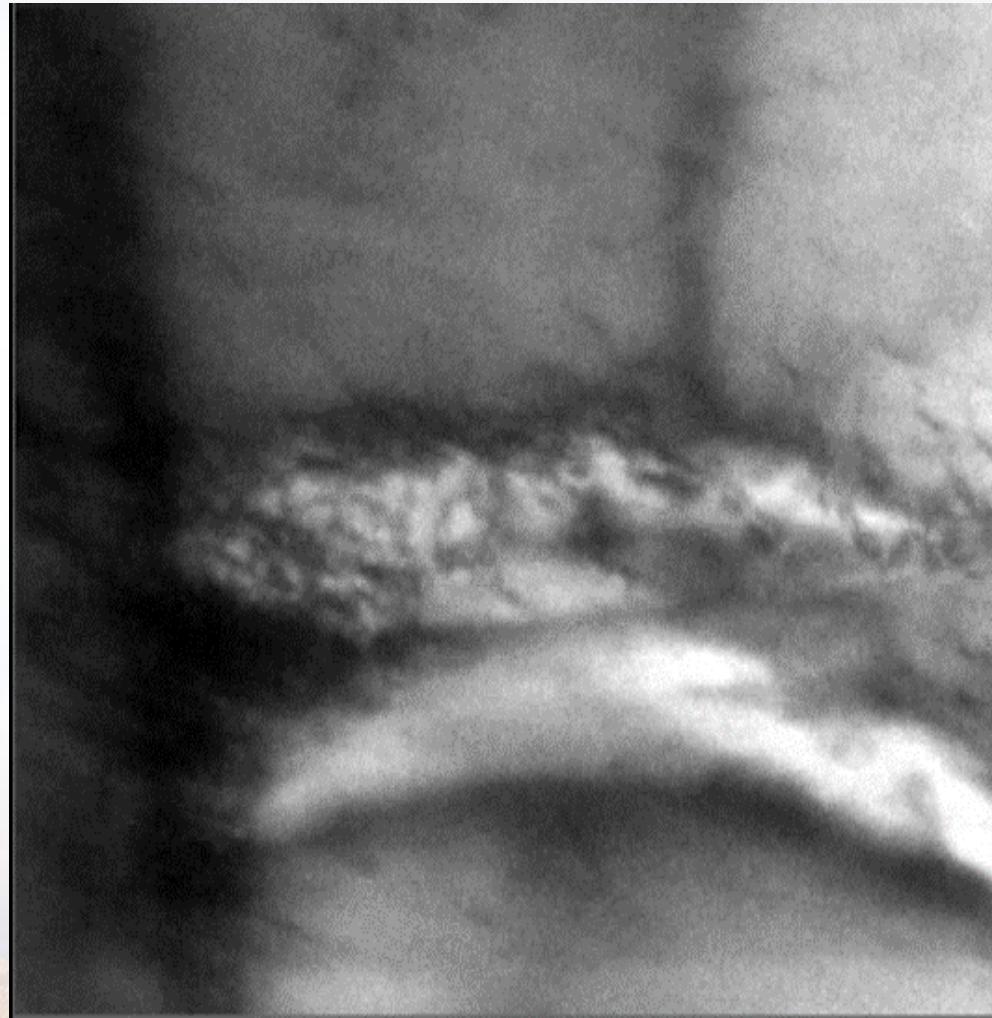
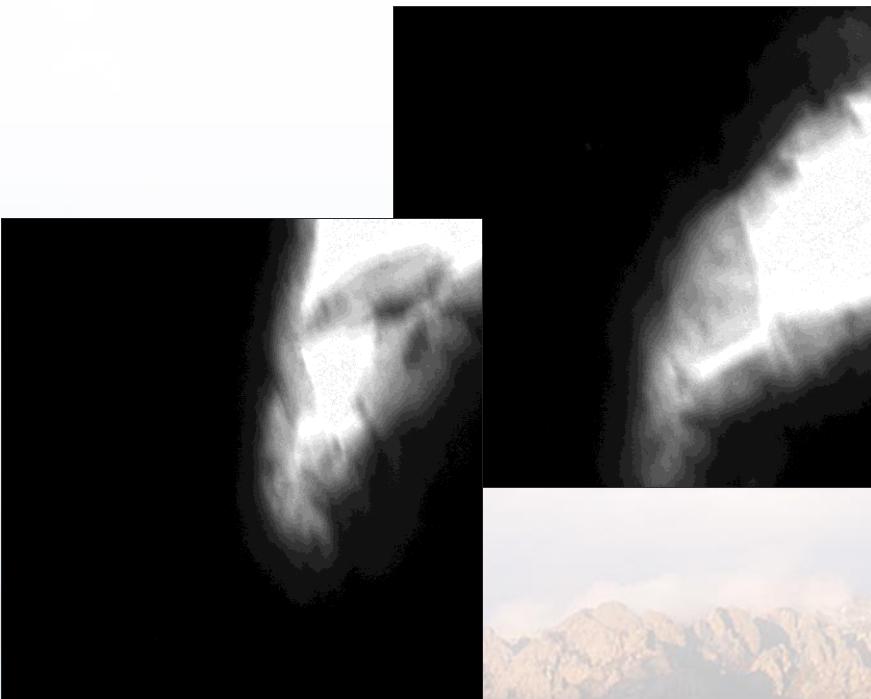


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In-situ D Irradiation

Pure Ni before and after Ni and D+ irradiation

- Mode of mechanical failure unchanged
- Dislocation movement



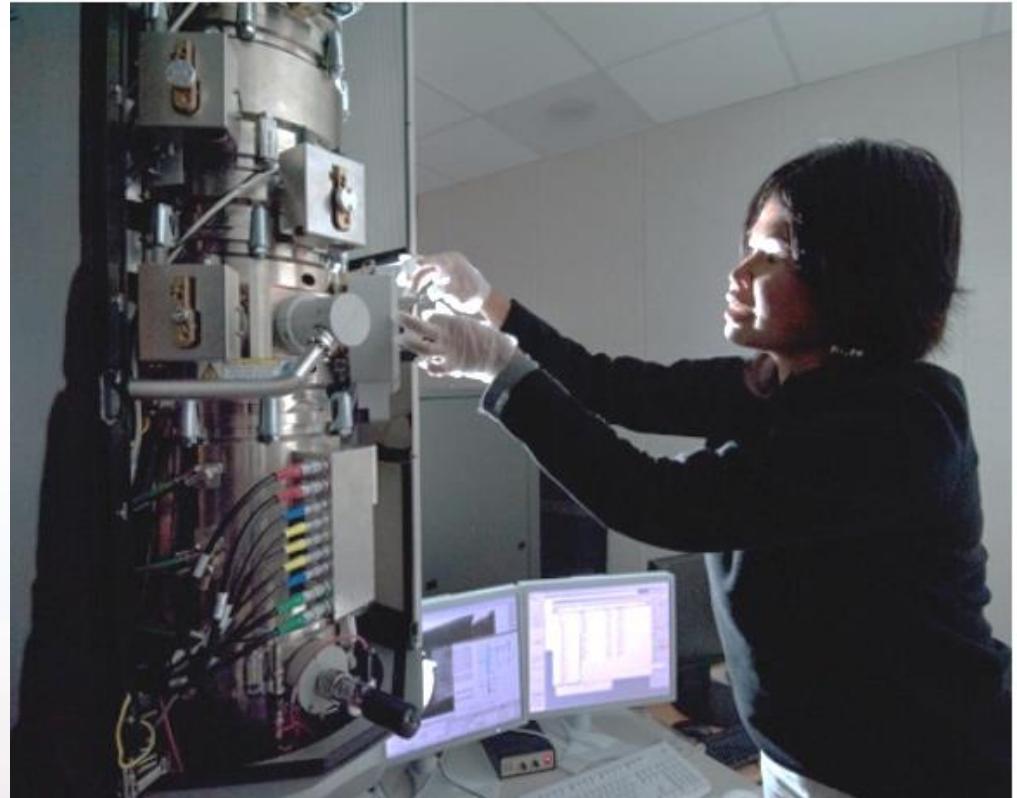
In-situ TEM techniques provide insight to mechanism of failure due to H embrittlement



In situ H₂ experiment at Stanford

- Imaging in H₂ gas using double-tilt heating holder
- Max pressure is ~10 mBar.
- Limited to lower pressure for hot-stage work due to thermal losses to gas (e.g, at 500° , P_{H2} max ~2-3 mBar).

- Aim: investigate predicted defaceting transition with H₂ exposure for $\Sigma=3$ boundaries in Nickel and the change in fracture mechanisms.

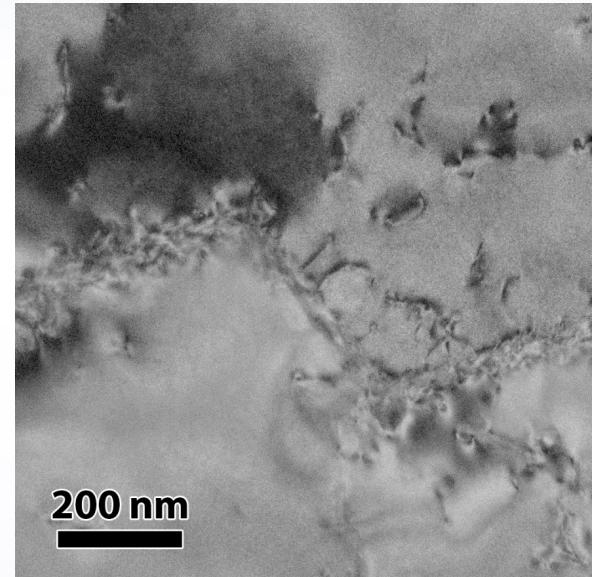
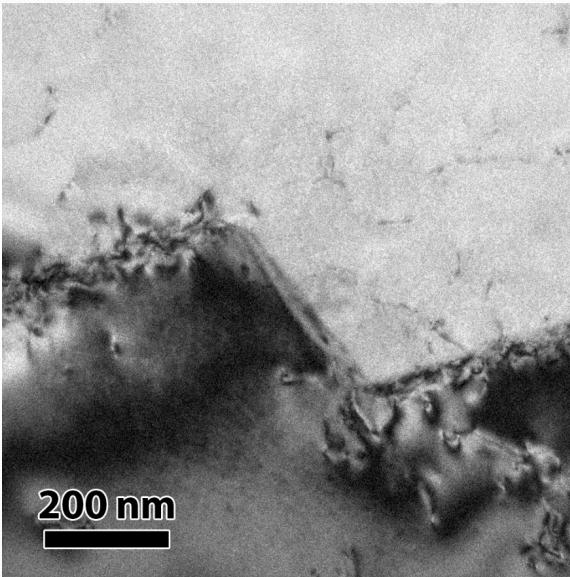
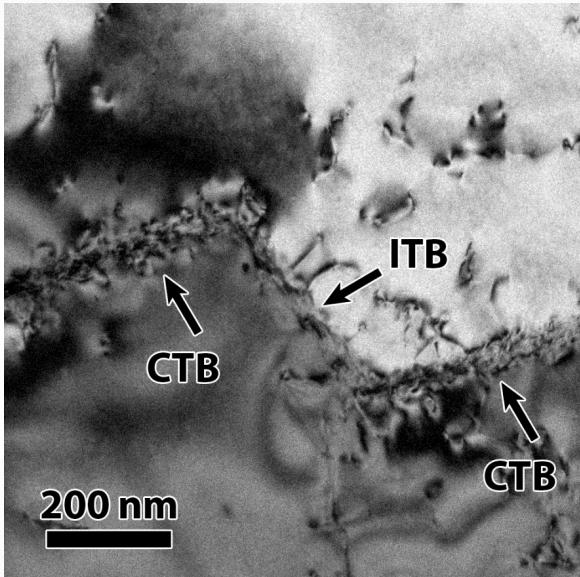


Environmental TEM at the Stanford Nano Shared Facility (SNSF).
(<http://snsf.stanford.edu/equipment/eim/titan.html>)



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In situ H₂ and Annealing Experiment



- Left to right: before, after, difference image
- Most notable changes: cleaning of ITB
- CTB/ITB corners may have rounded, but some of that may be to tilt. Disappearance of dislocations in upper portion is due to change in imaging condition.

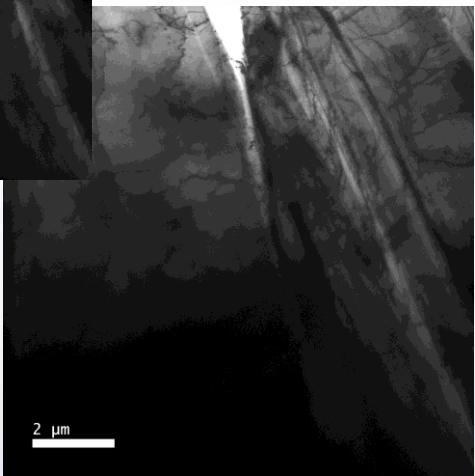
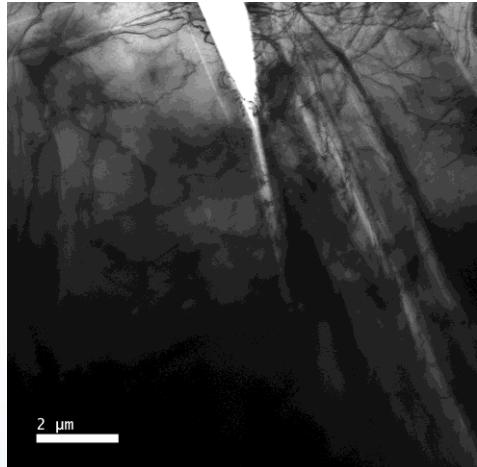


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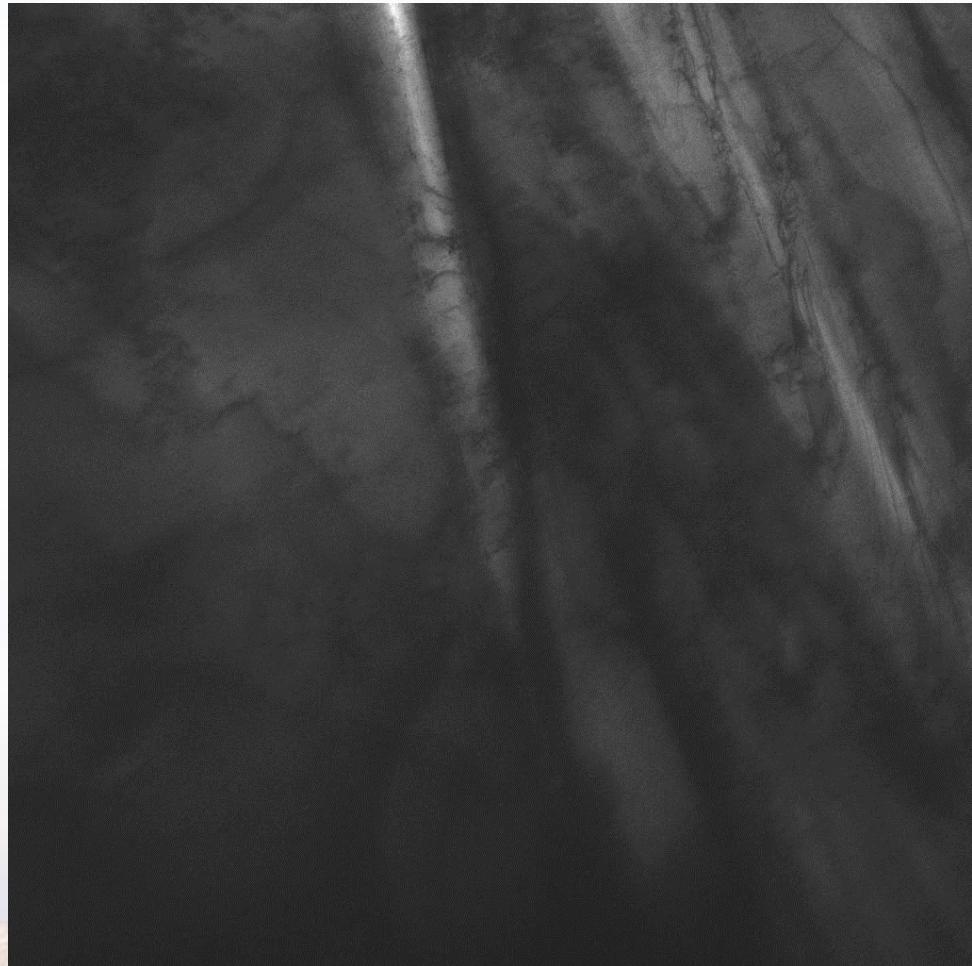
In situ H₂ and Straining Experiment

Ni behavior during H flow

- Dislocation behavior during H introduction
 - Between 10⁻³ and 9 mbar



Pre-existing crack in pure Ni at 10⁻³mbar (left) and in a 9mbar H environment (right)



In-situ TEM techniques provide insight to mechanism of failure due to H embrittlement



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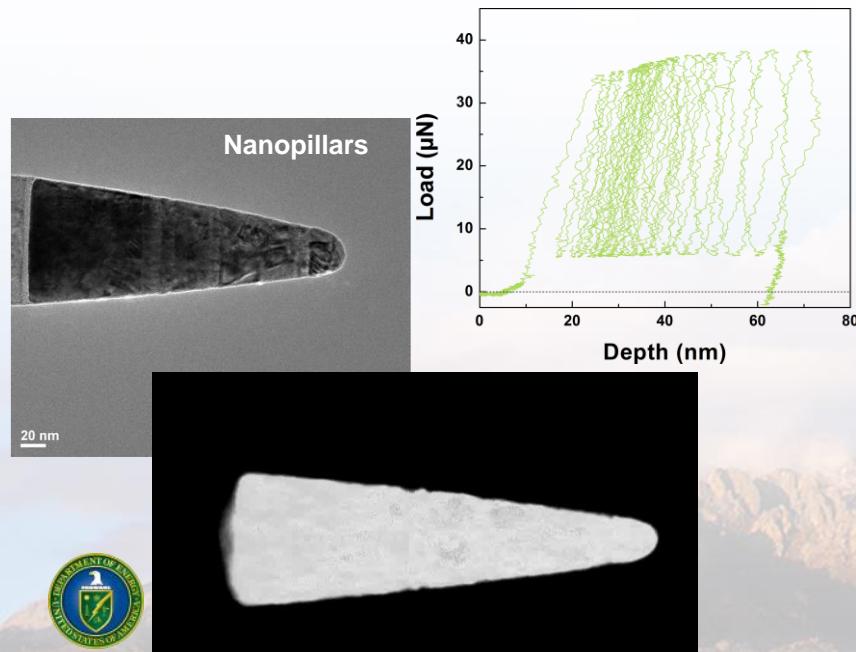
Future Directions

In situ Quantitative Mechanical Testing



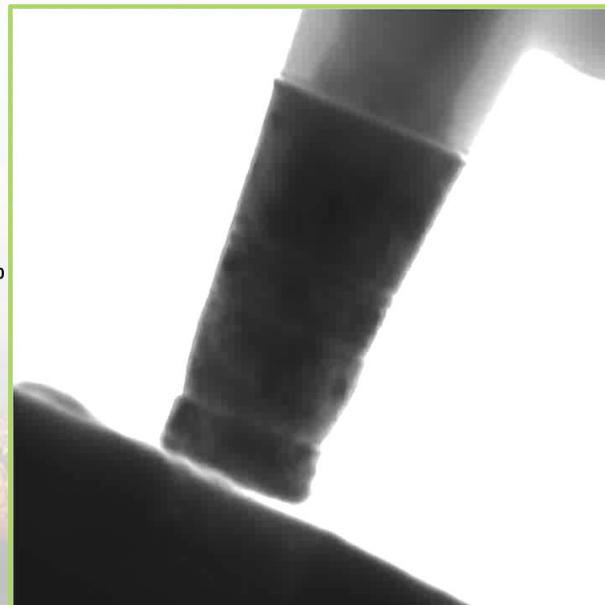
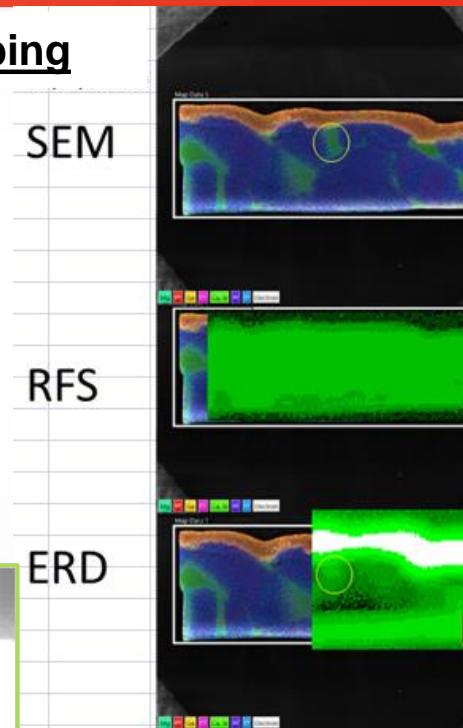
Hysitron PI95 *In Situ* Nanoindentation TEM Holder

- Sub nanometer displacement resolution
- Quantitative force information with μN resolution
- **Concurrent real-time imaging by TEM**



Hydrogen Isotope Mapping

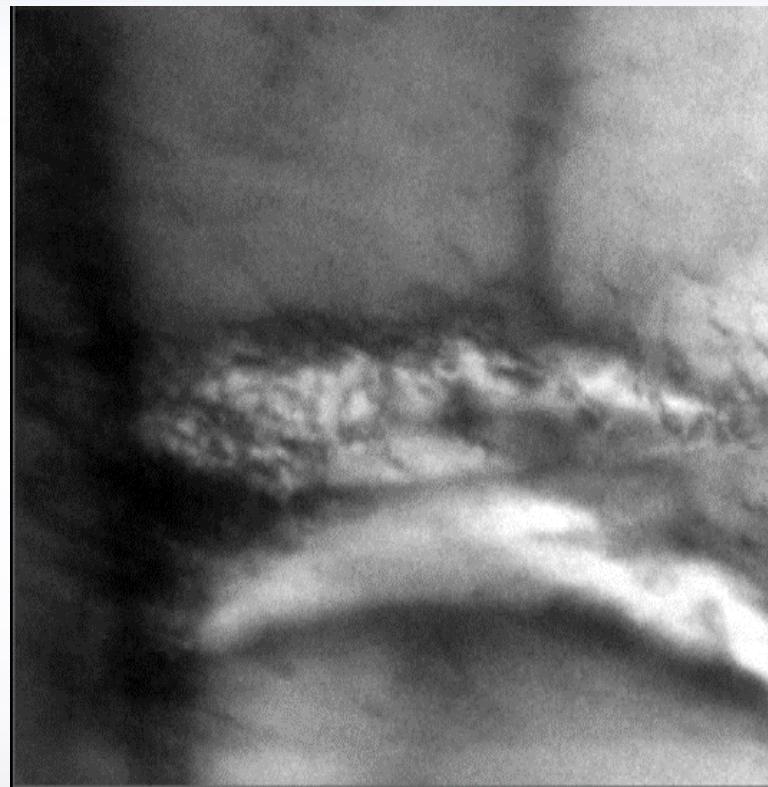
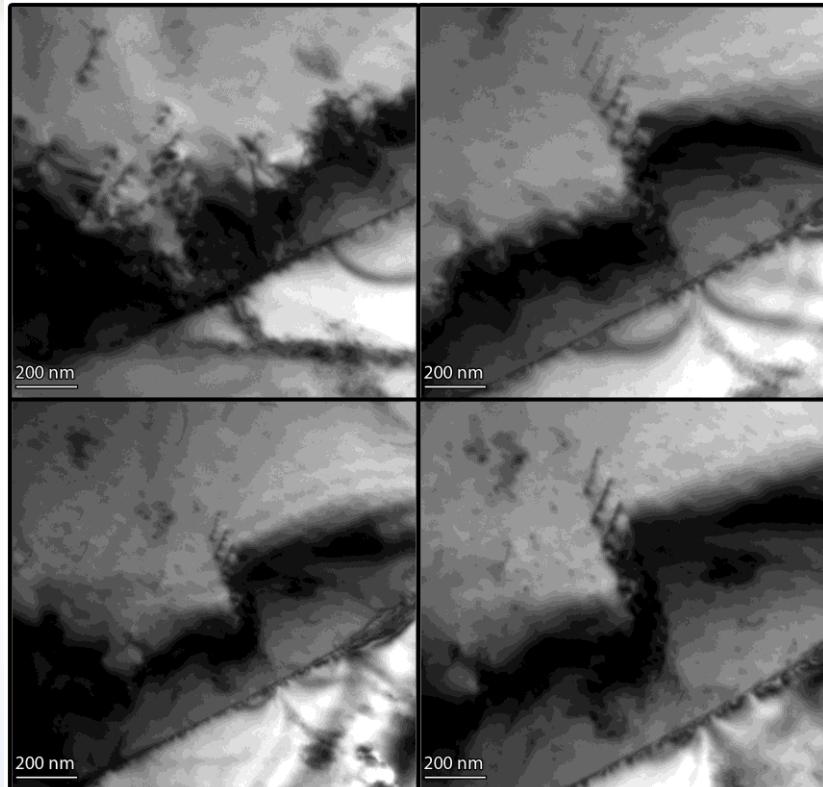
Development of methods to chemically map out hydrogen and helium isotopes using transmission electron elastic recoil detection



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Summary



In situ hydrogen gas exposure appears to be the best path to study defect response to hydrogen environment



External access and collaborations available through CINT and NSUF user proposals

Please contact Khalid Hattar khattar@sandia.gov for more information

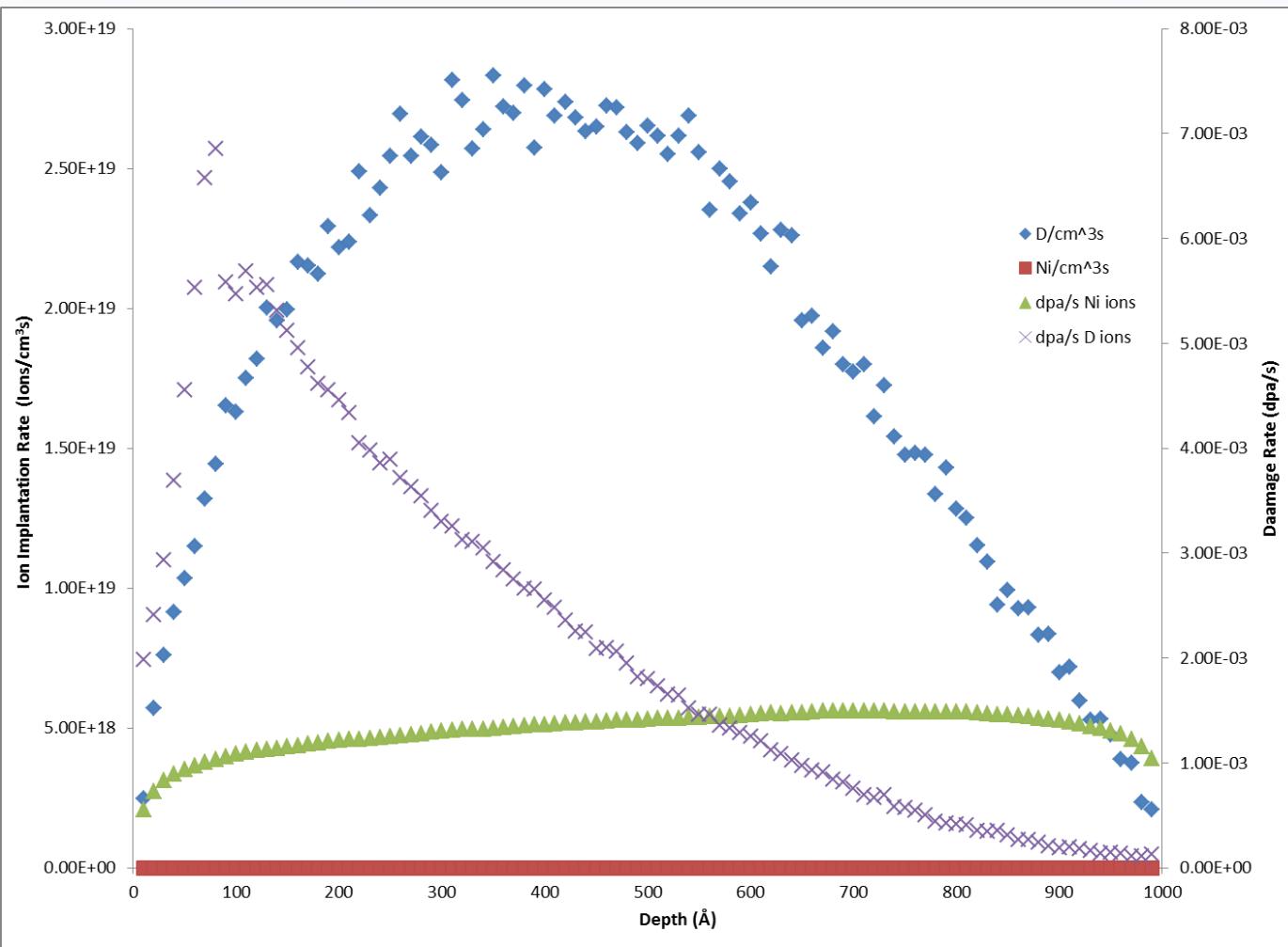
This work was supported by the US Department of Energy, Office of Basic Energy Sciences. Or other funding source.

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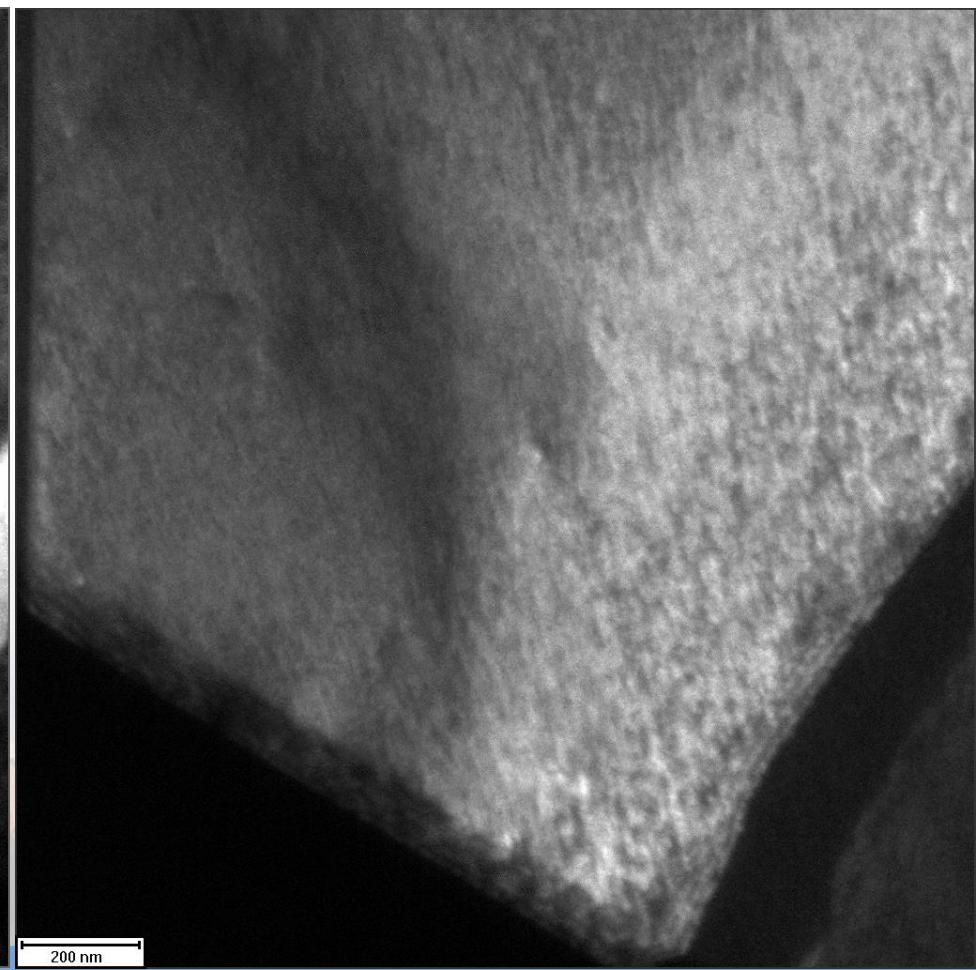
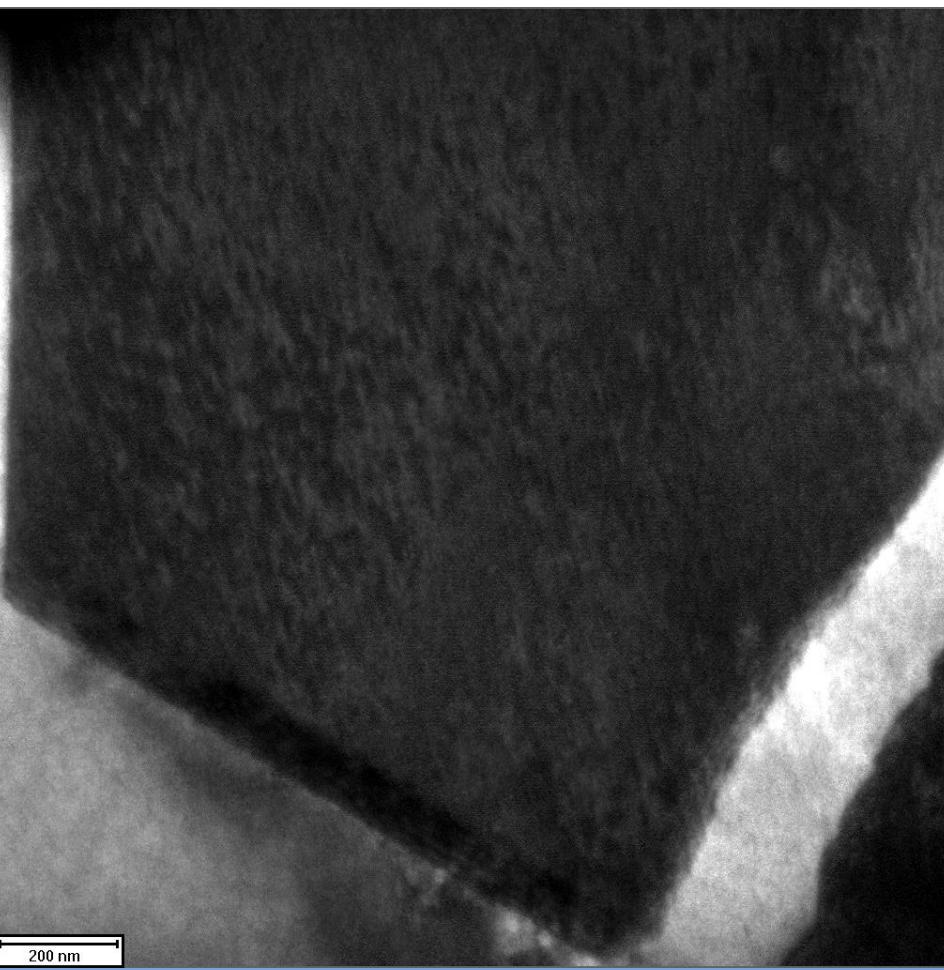
D⁺ and Ni³⁺ into large grained Jet polished Nickel



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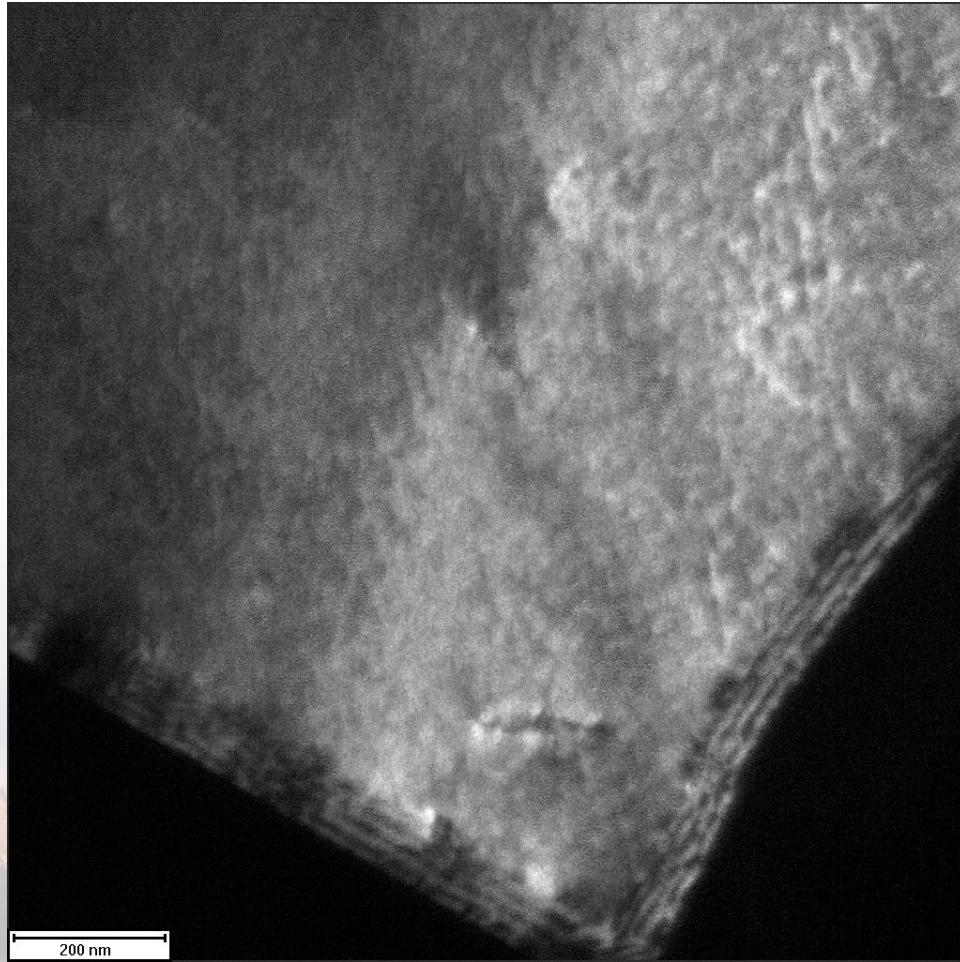
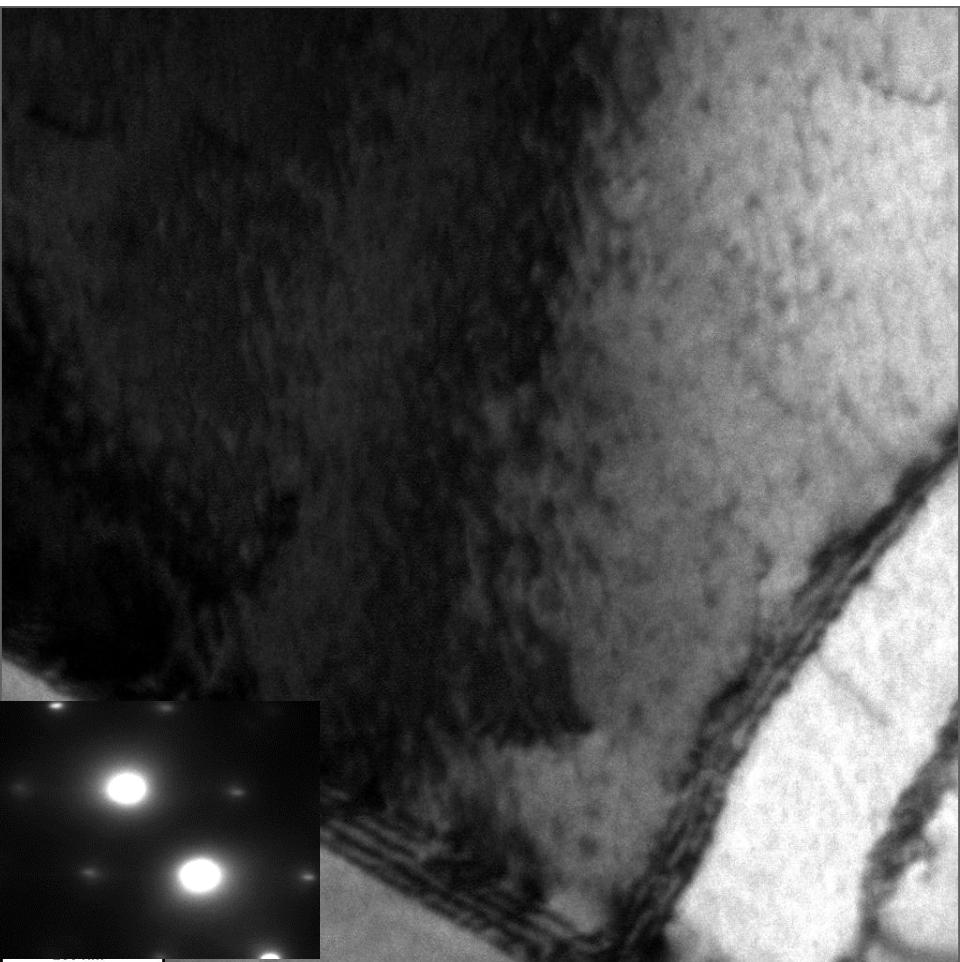
After D⁺ implantation





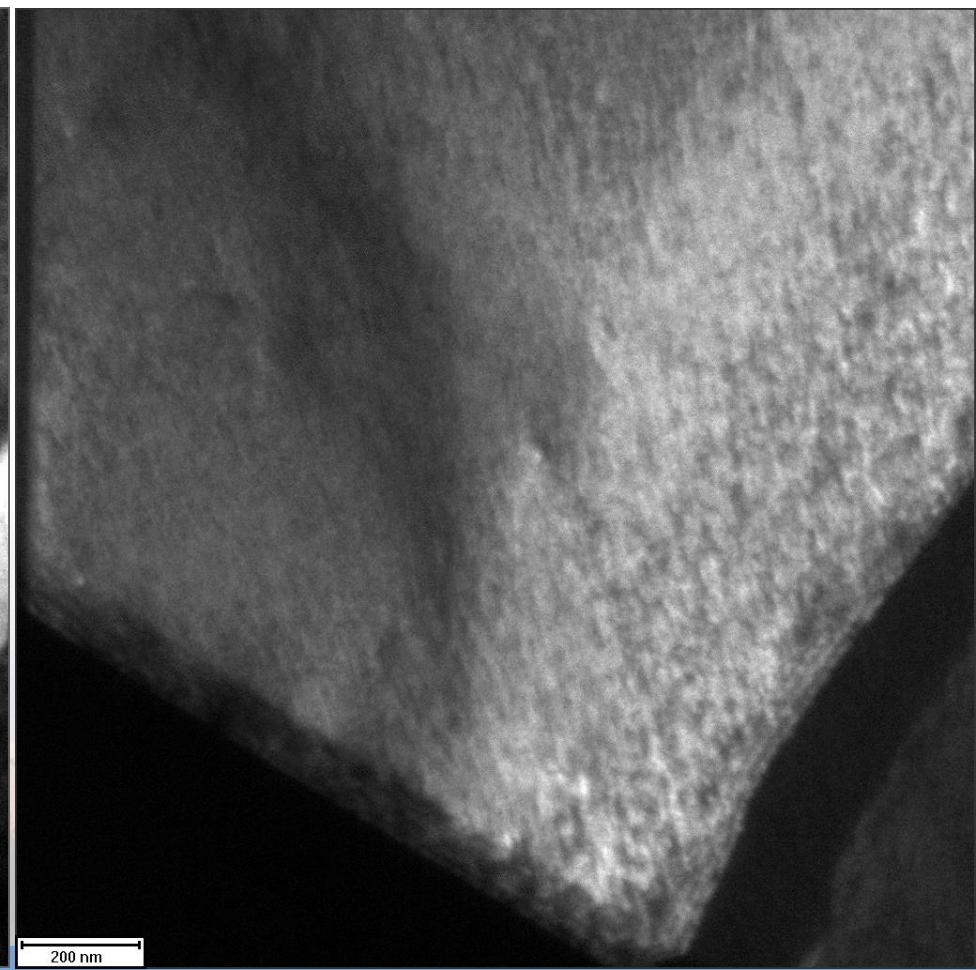
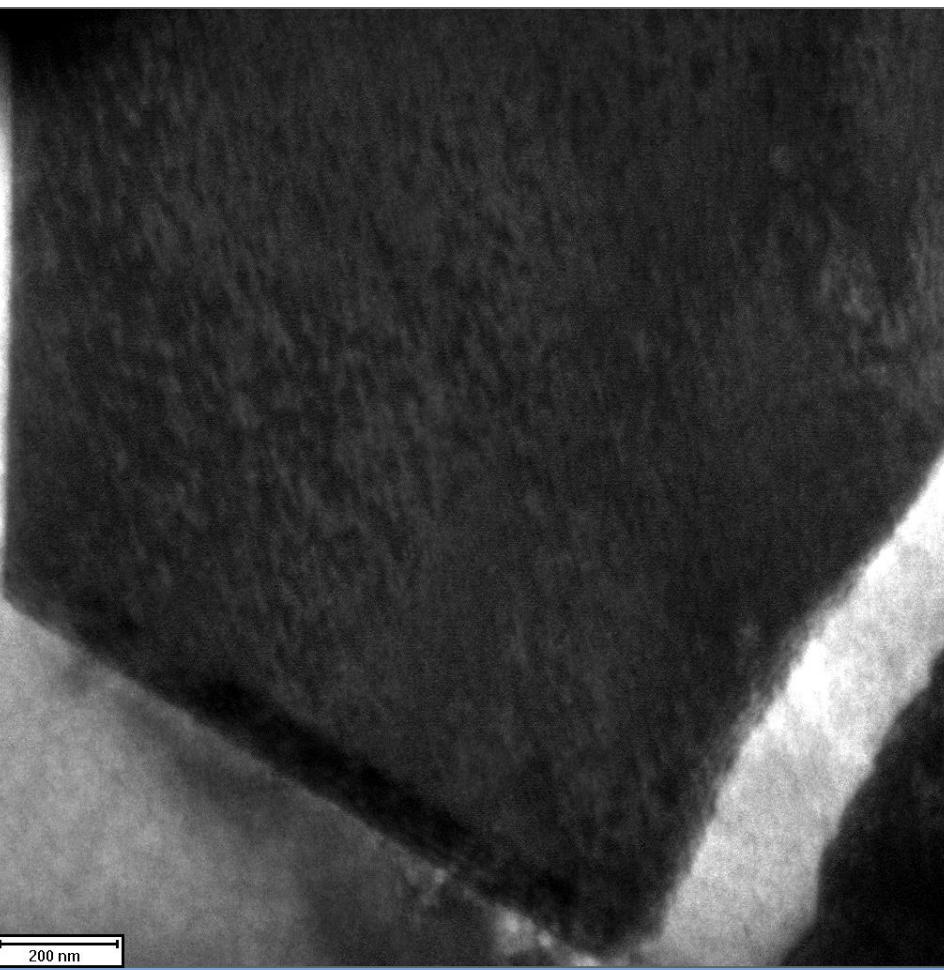
1: D⁺ then Ni³⁺

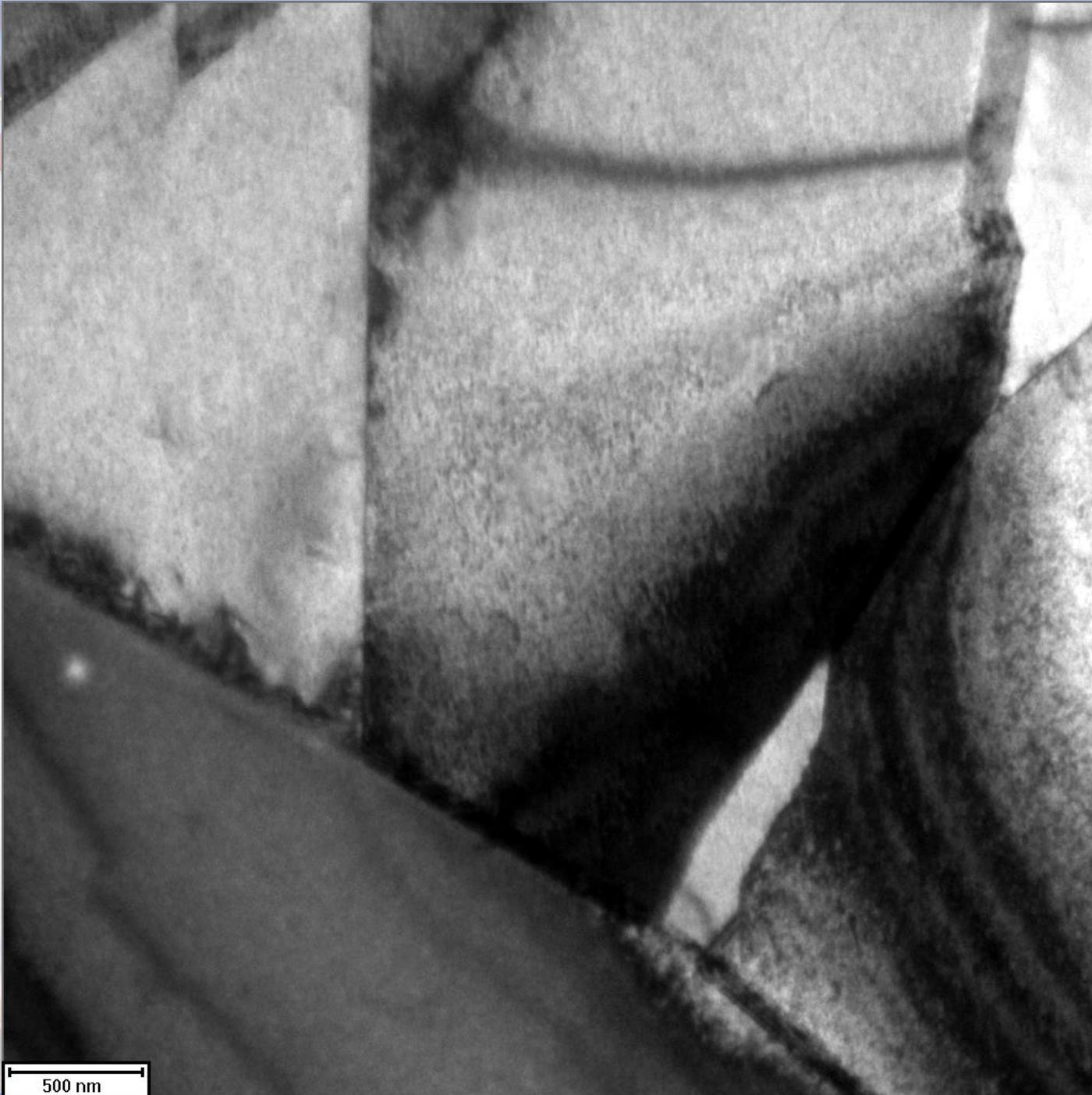
Before irradiation





After D⁺ implantation





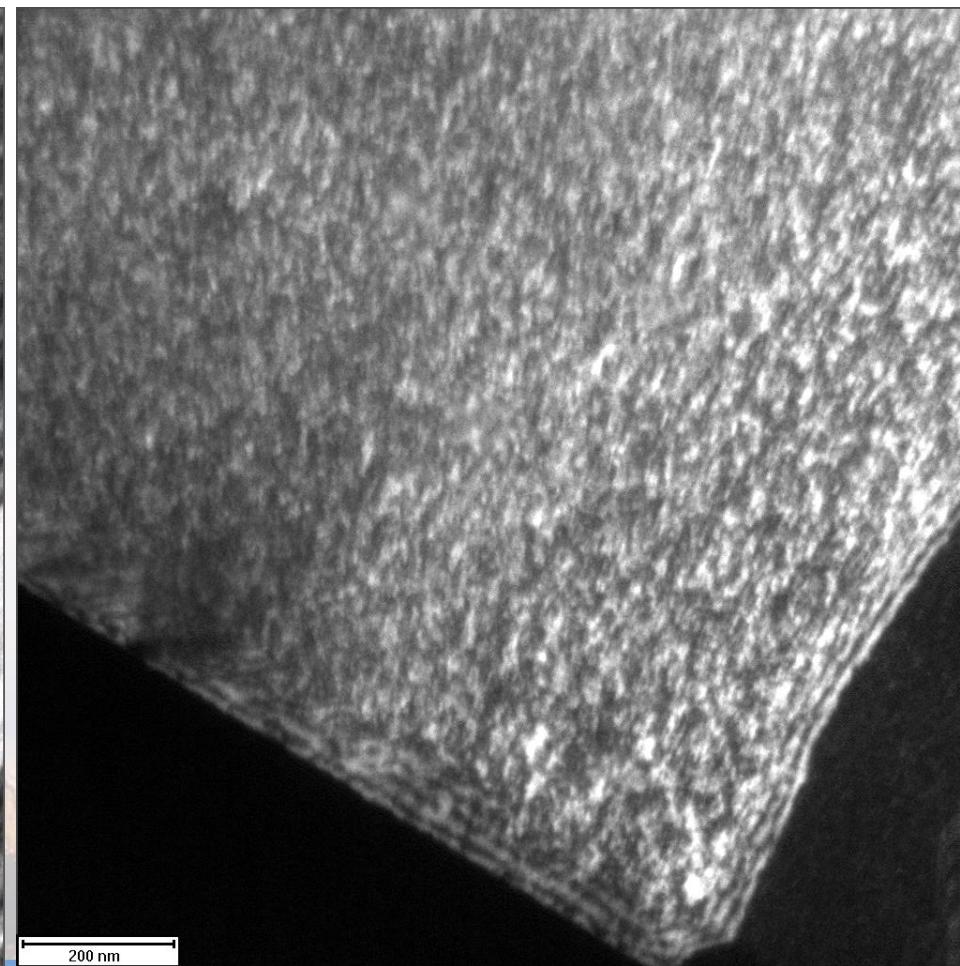
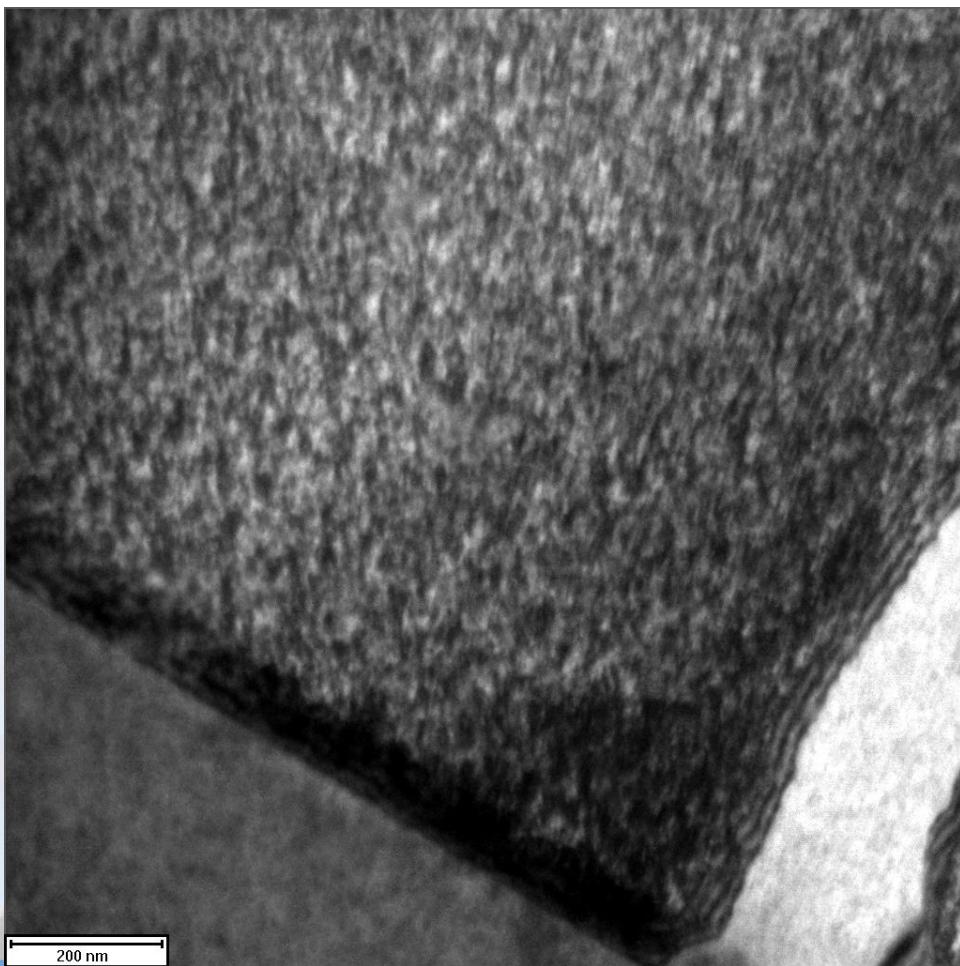
500 nm

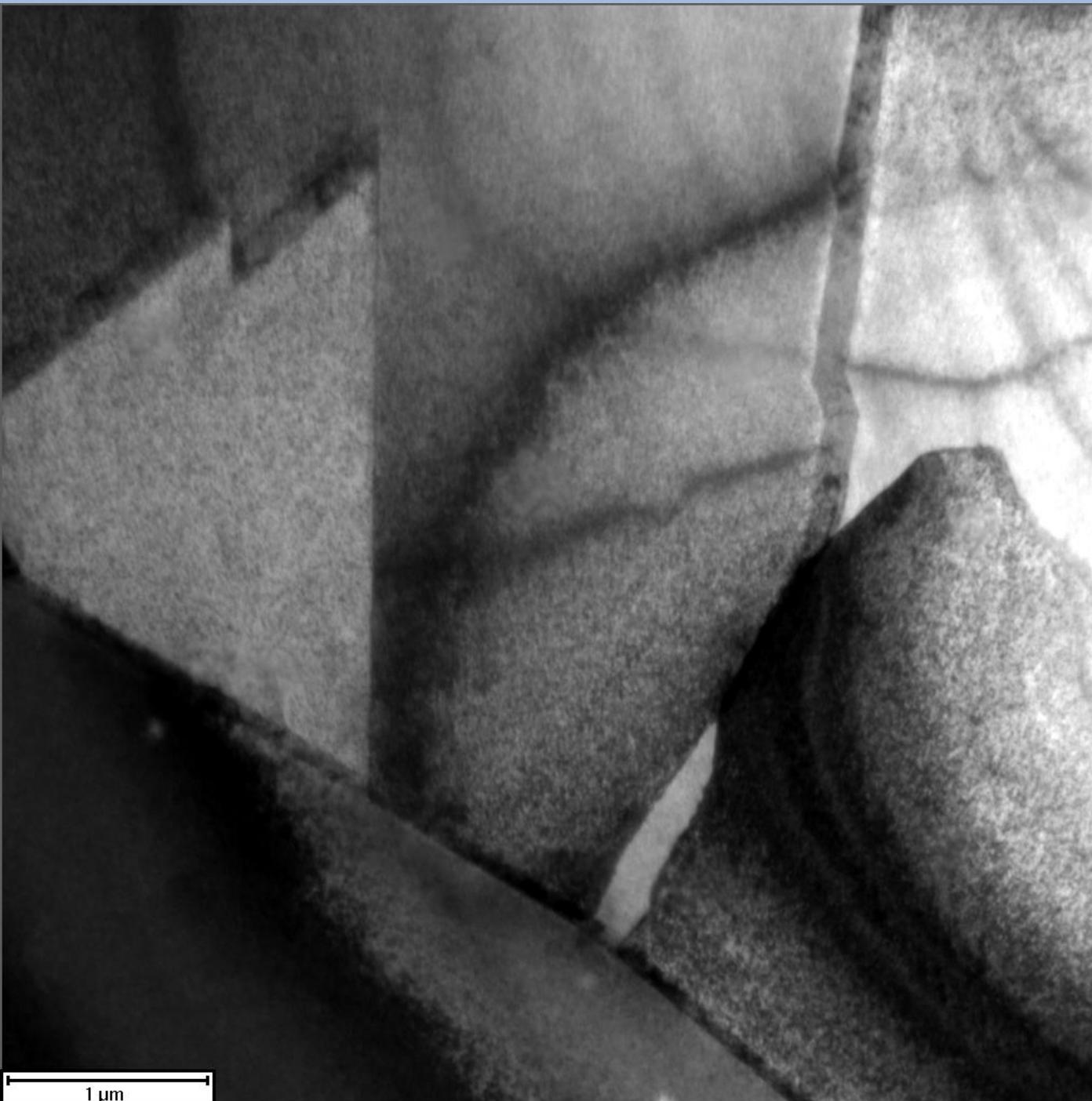


Laboratories



After Ni^{3+} additional irradiation

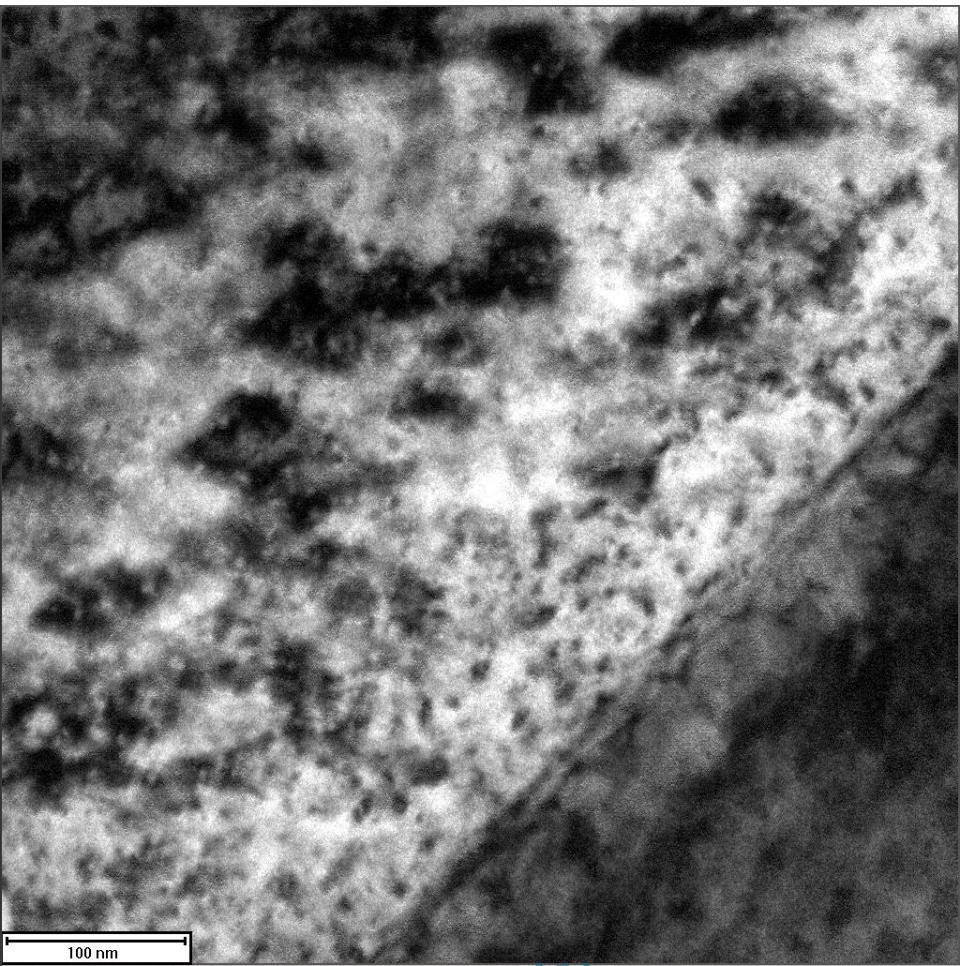
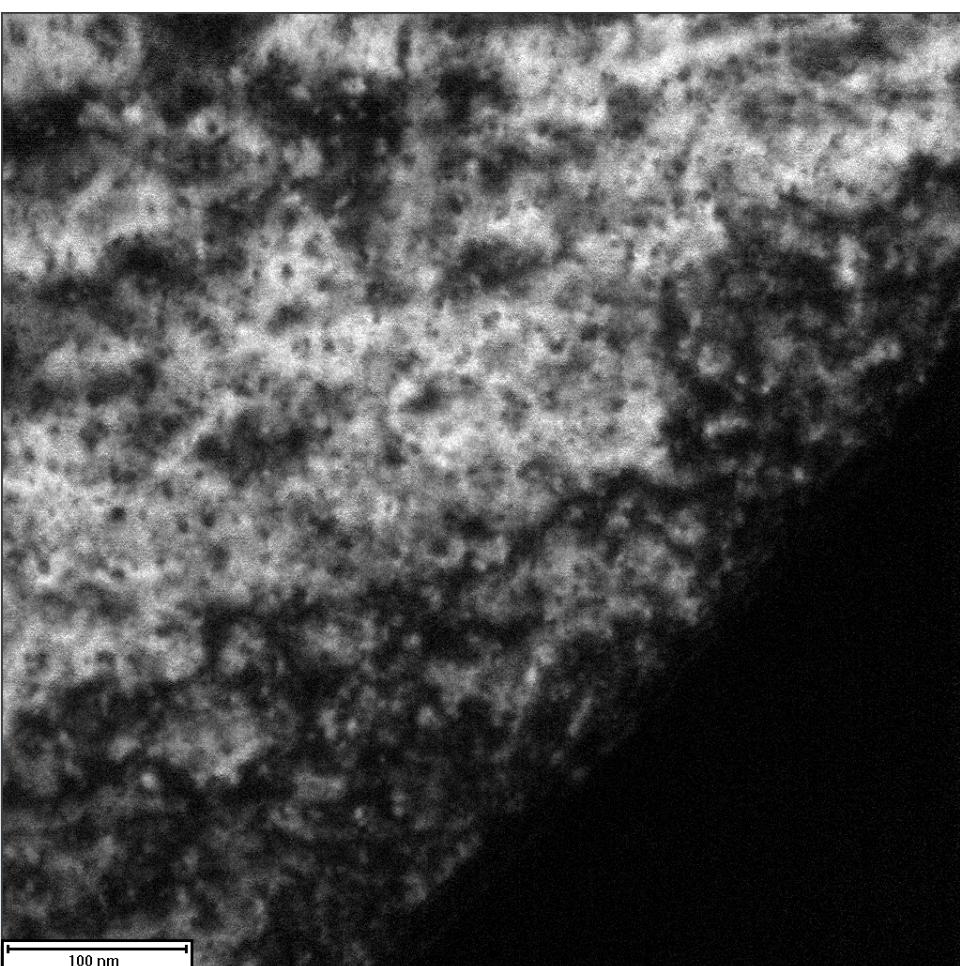




1 μ m



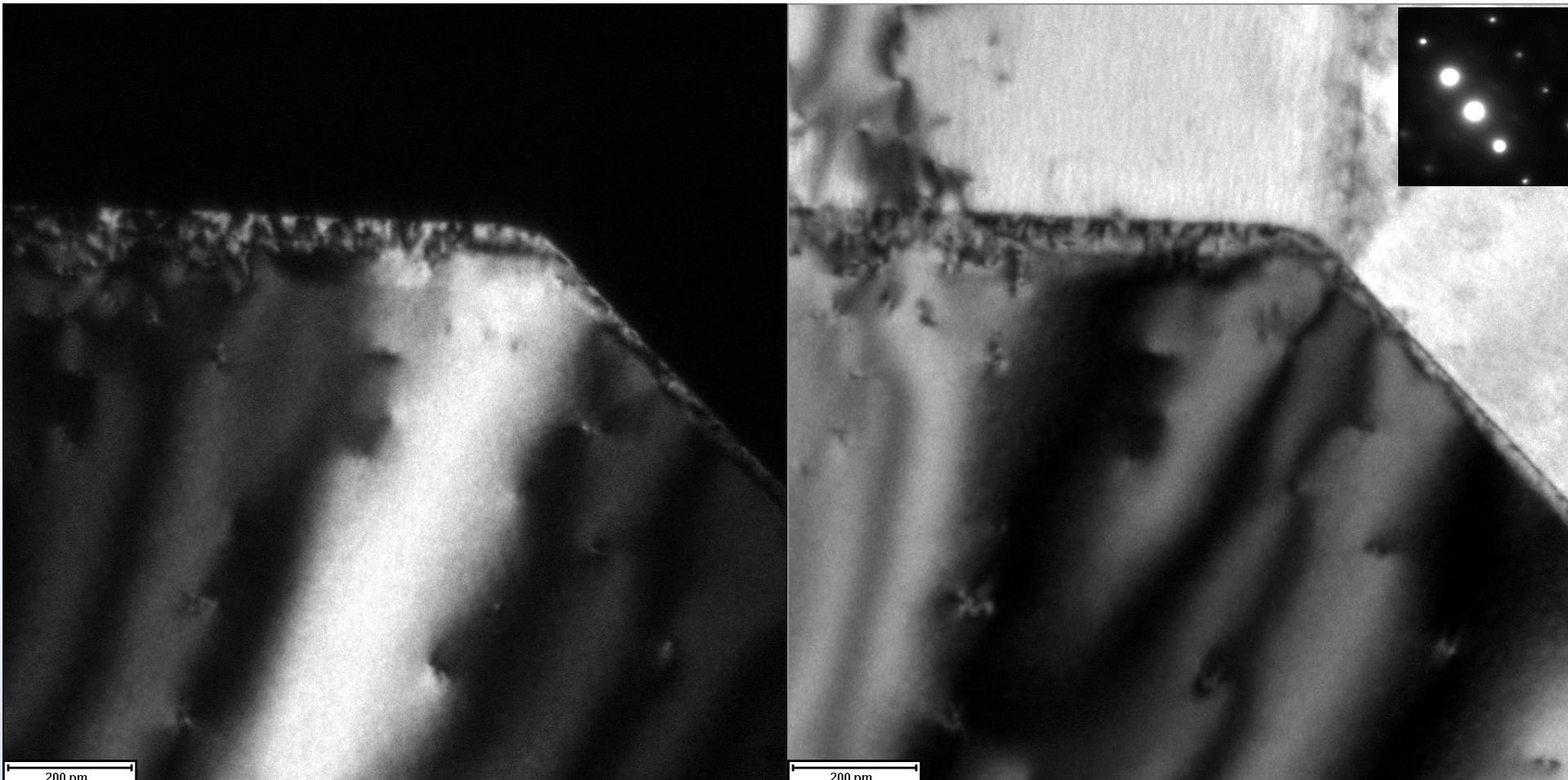
aboratories





#2: Ni³⁺ then D⁺

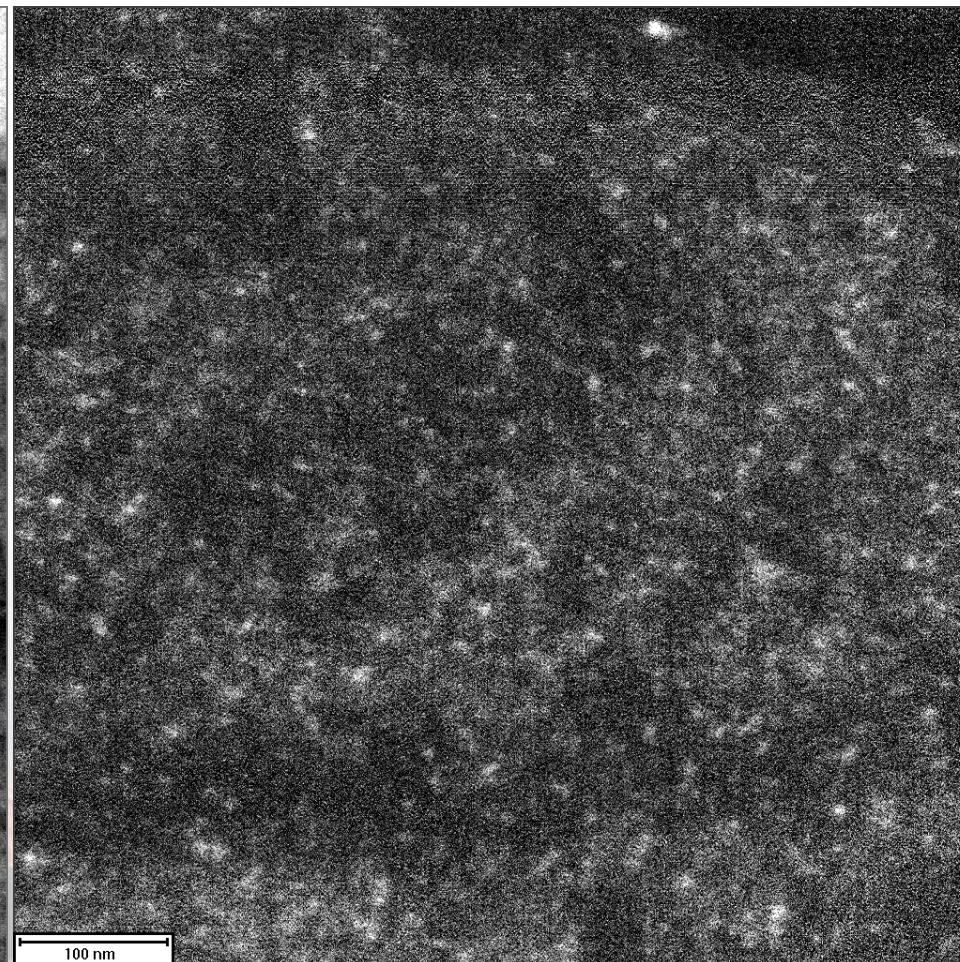
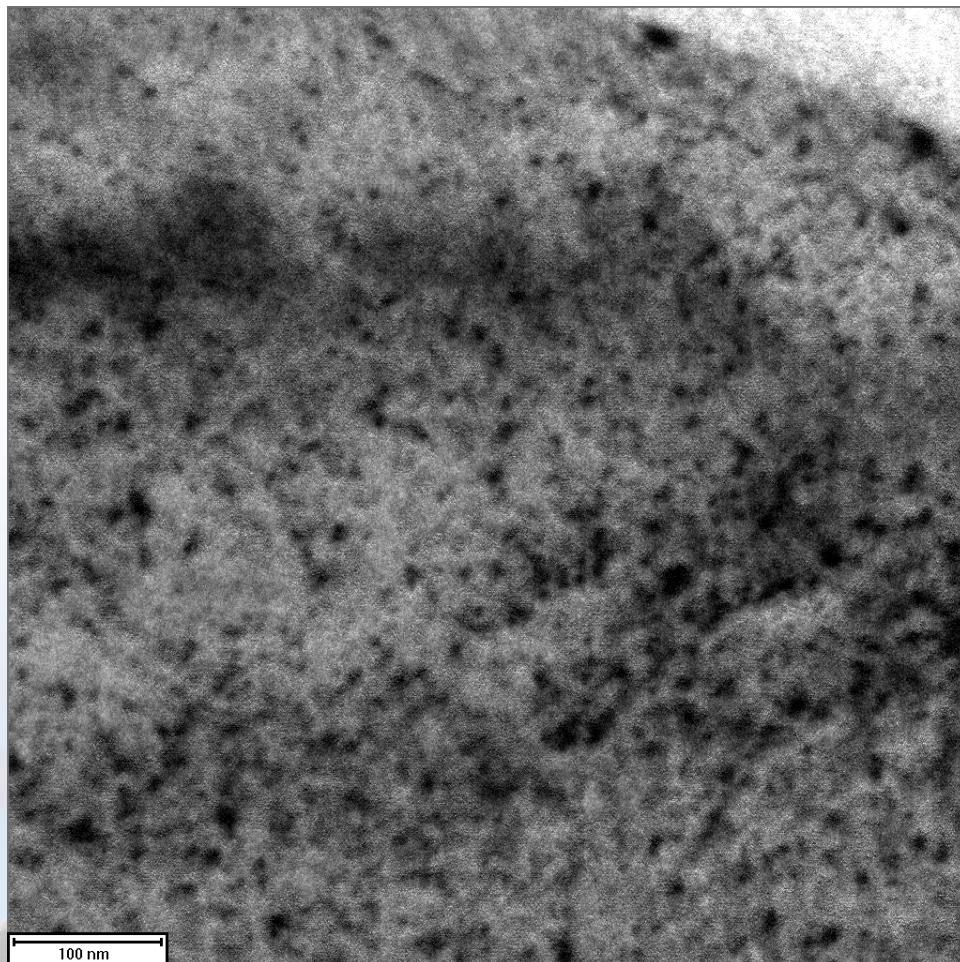
Before Irradiation

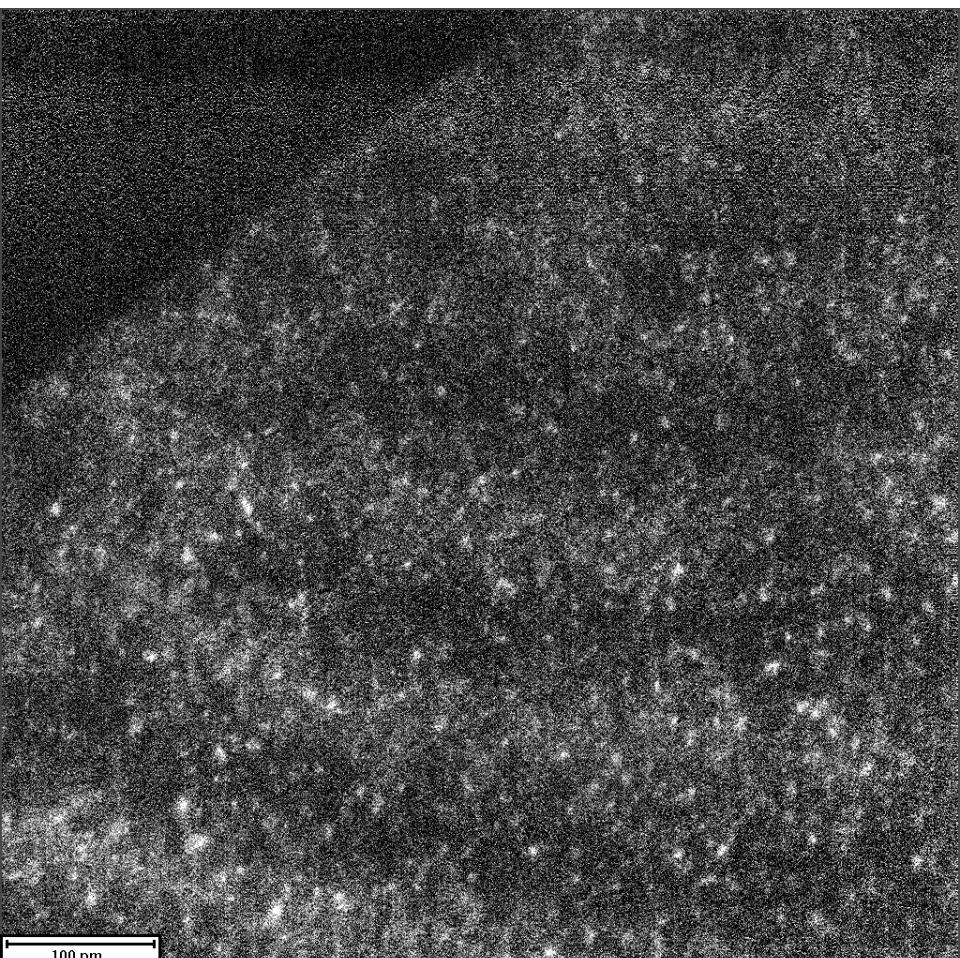
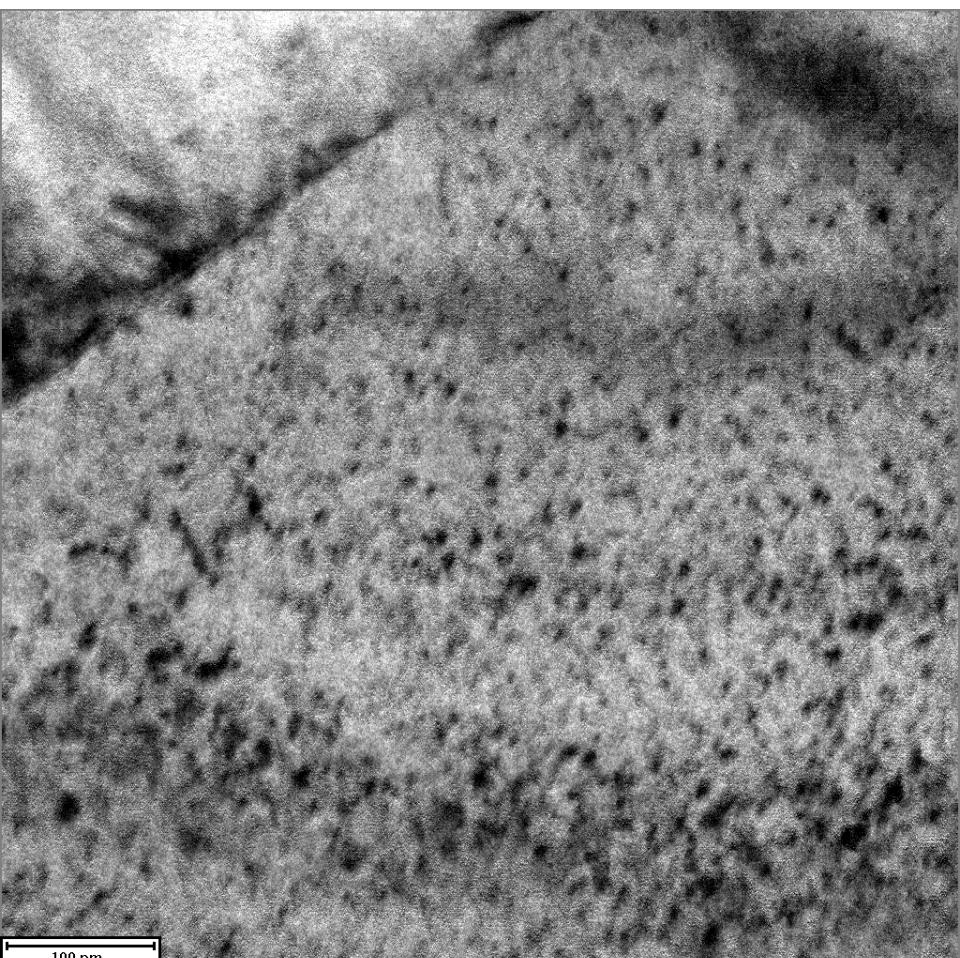


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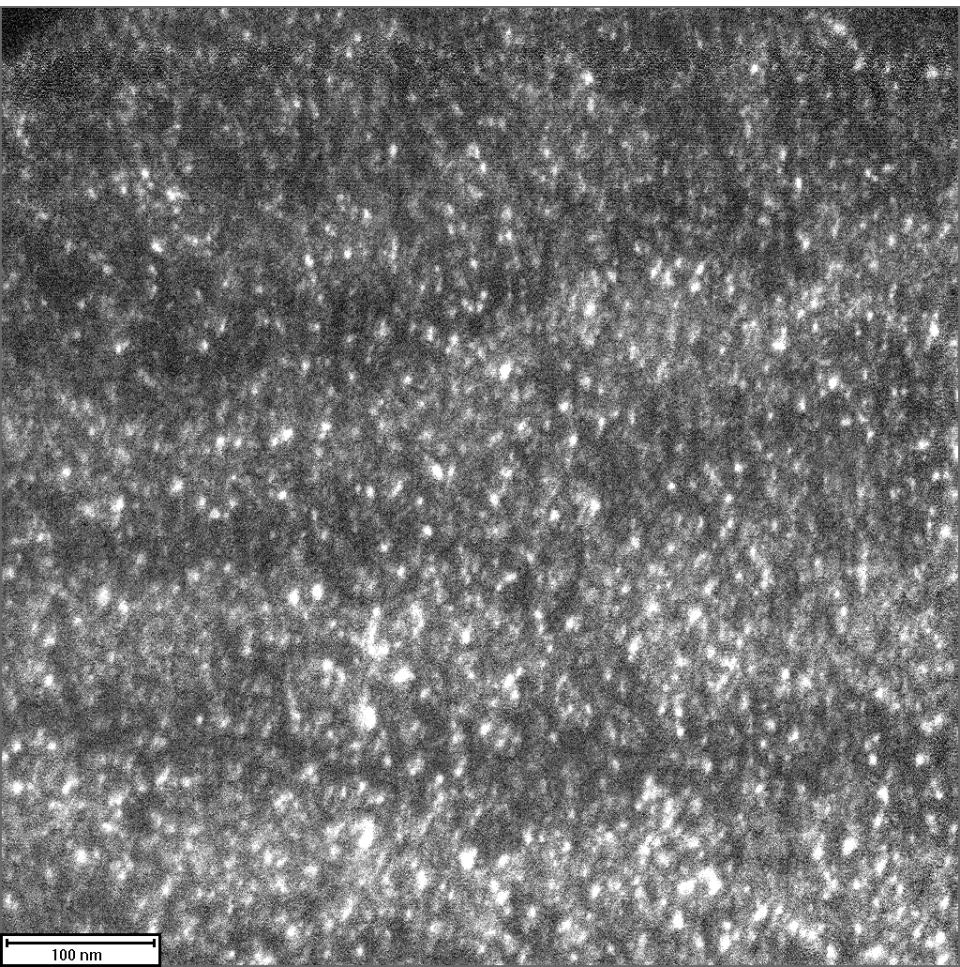
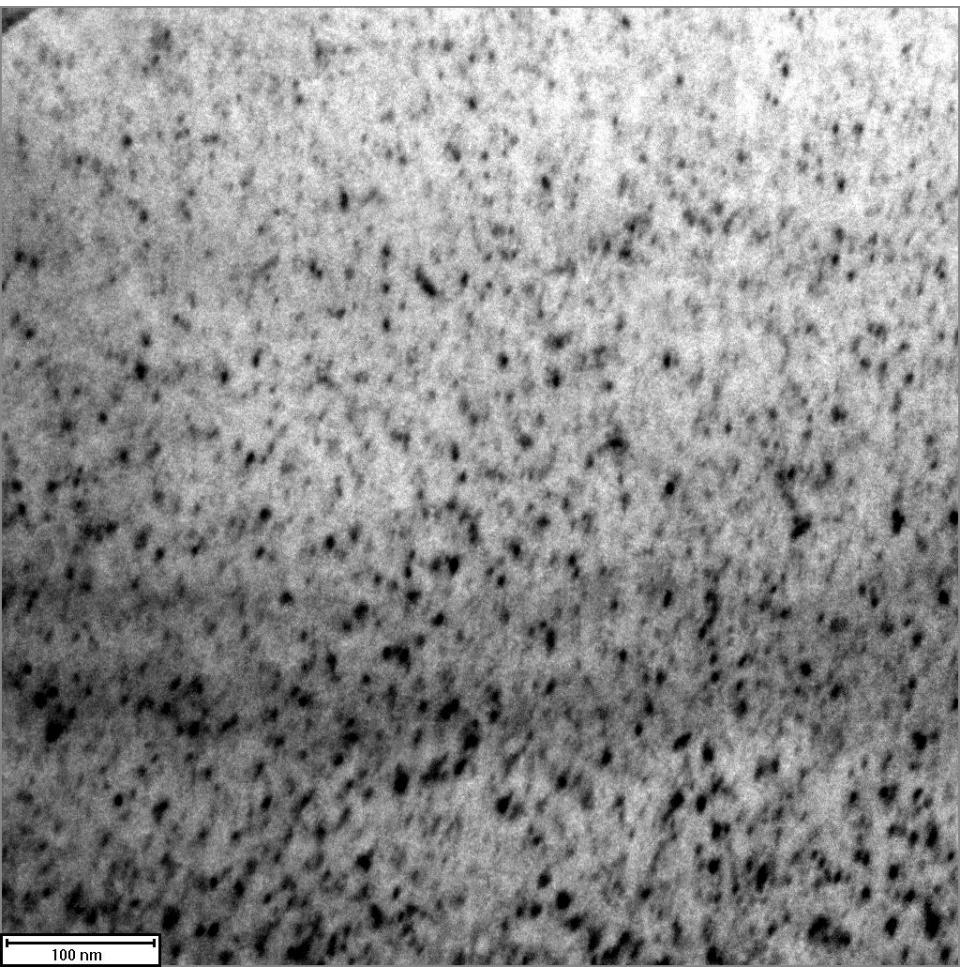
After Ni^{3+}





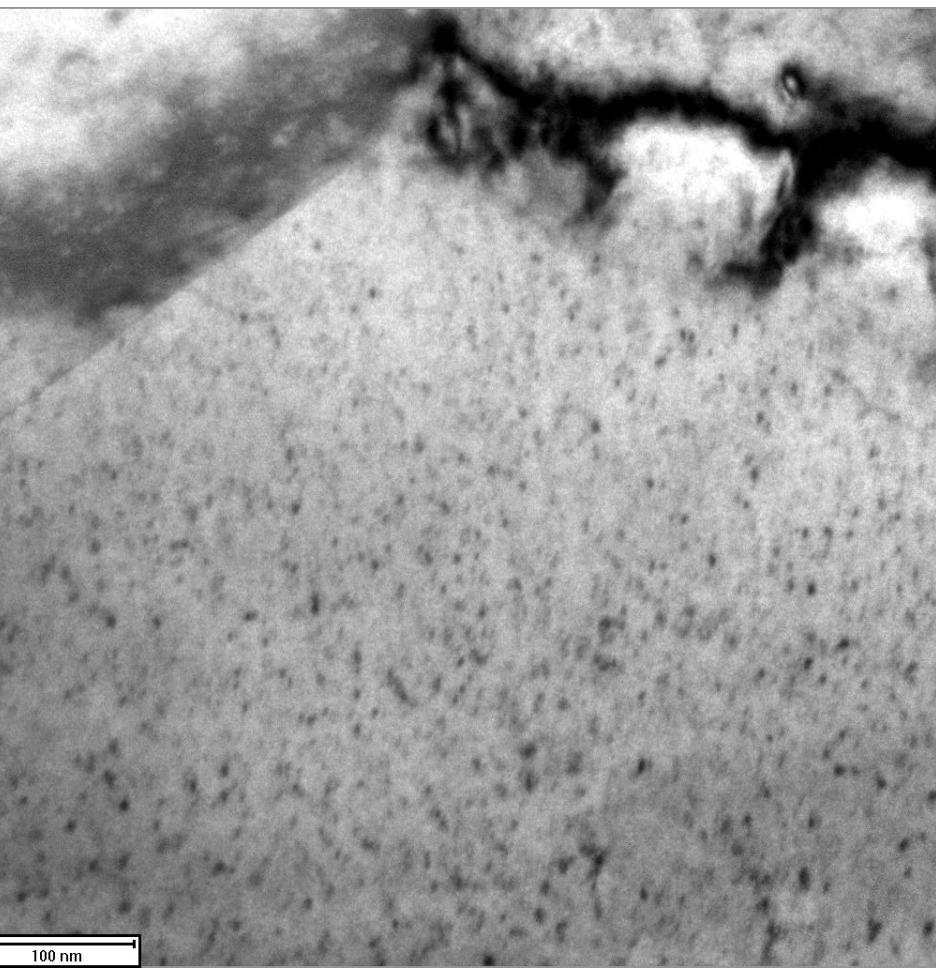


After addition of D⁺

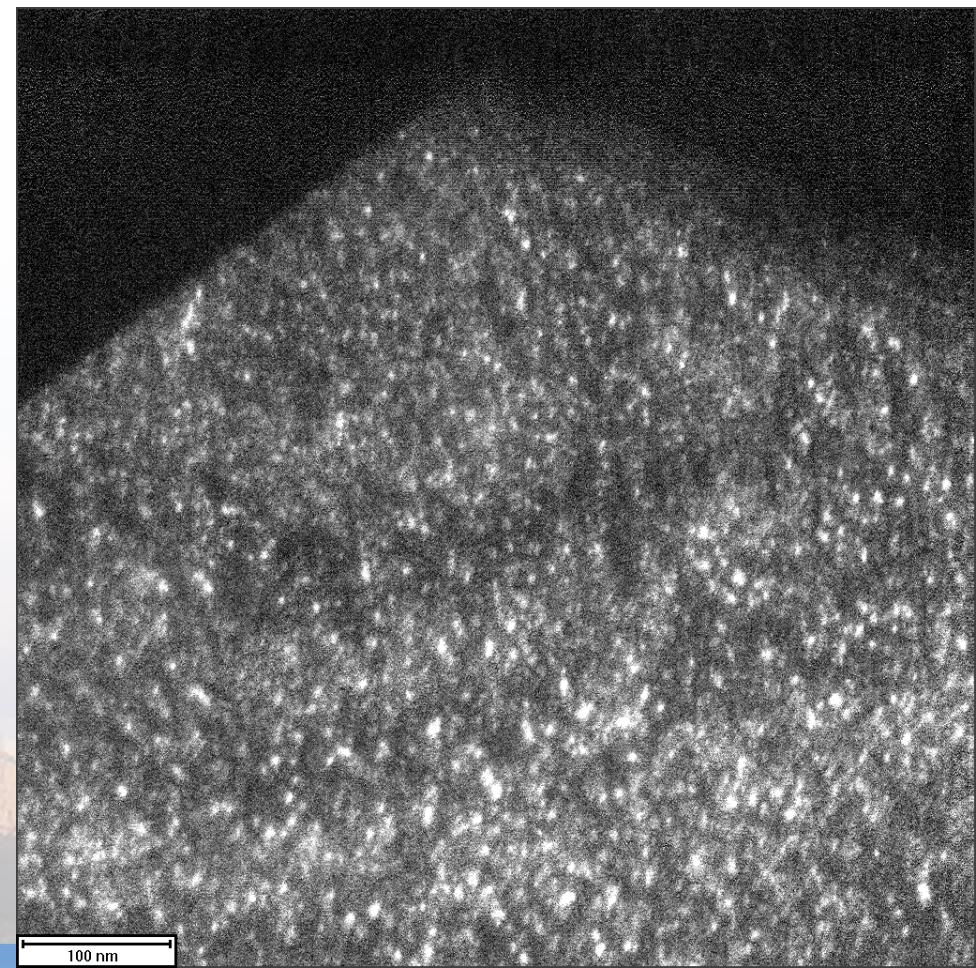


100 nm

100 nm



100 nm

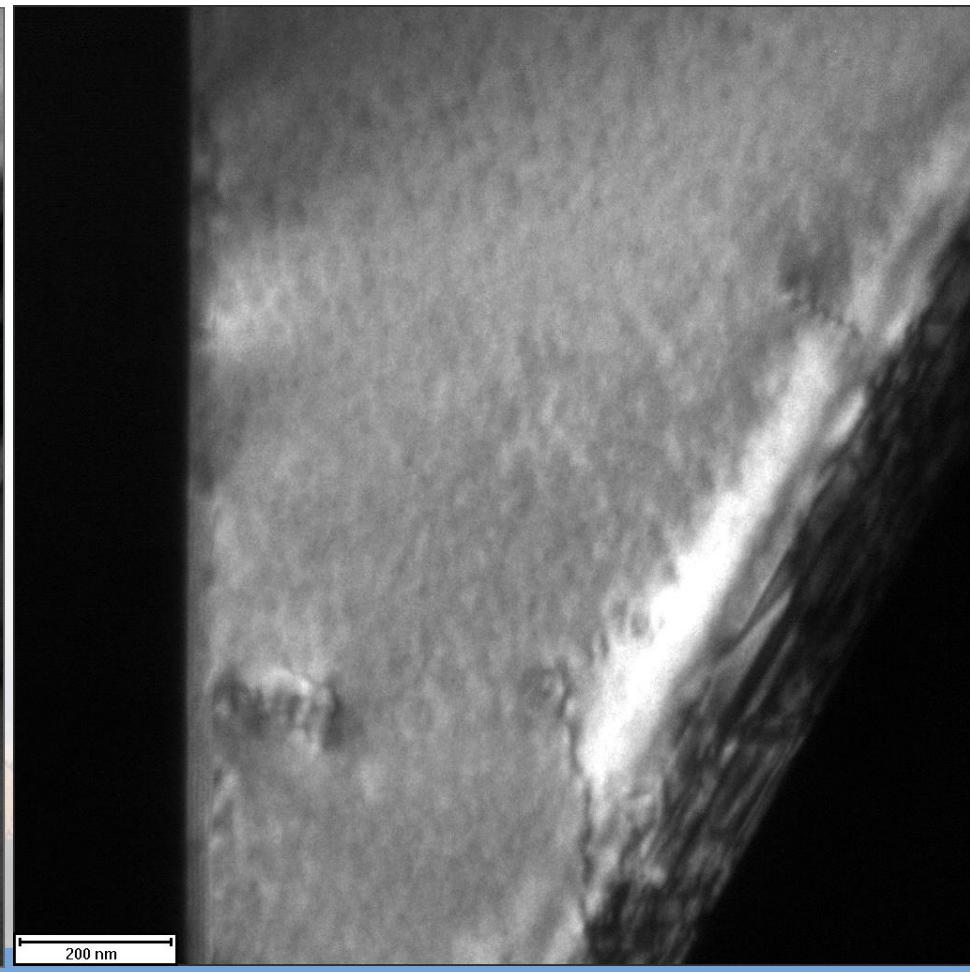
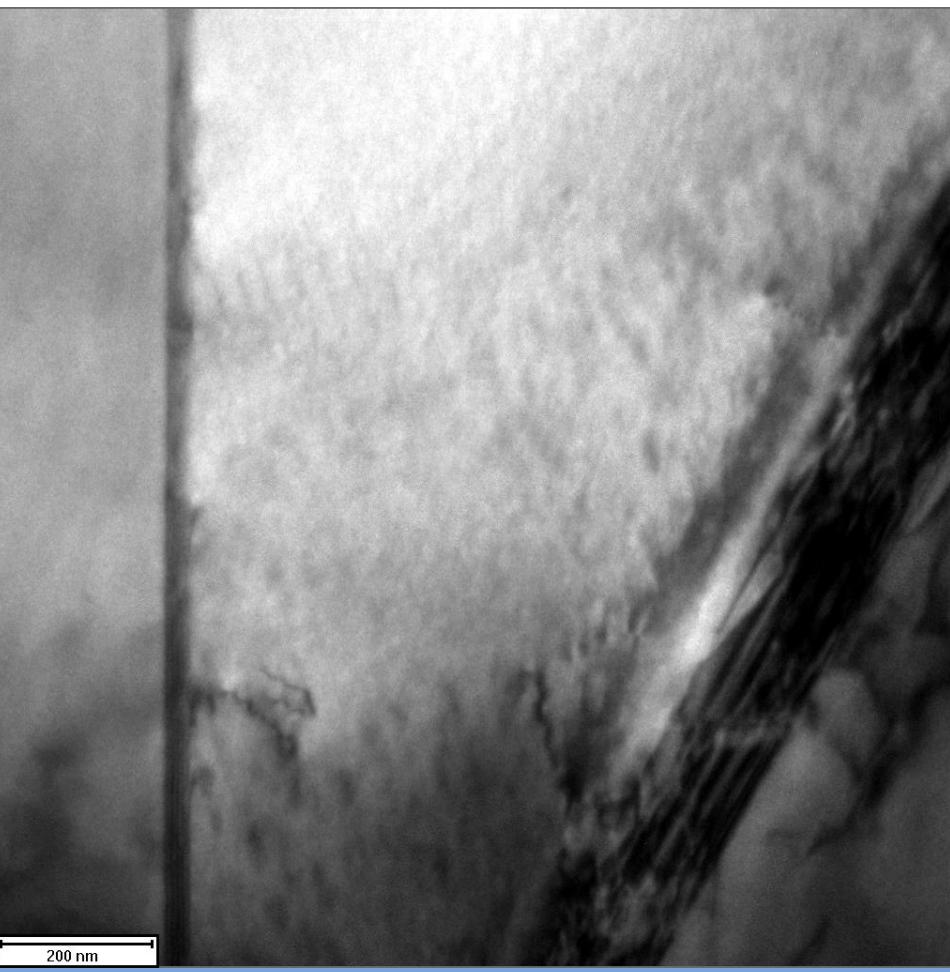


100 nm



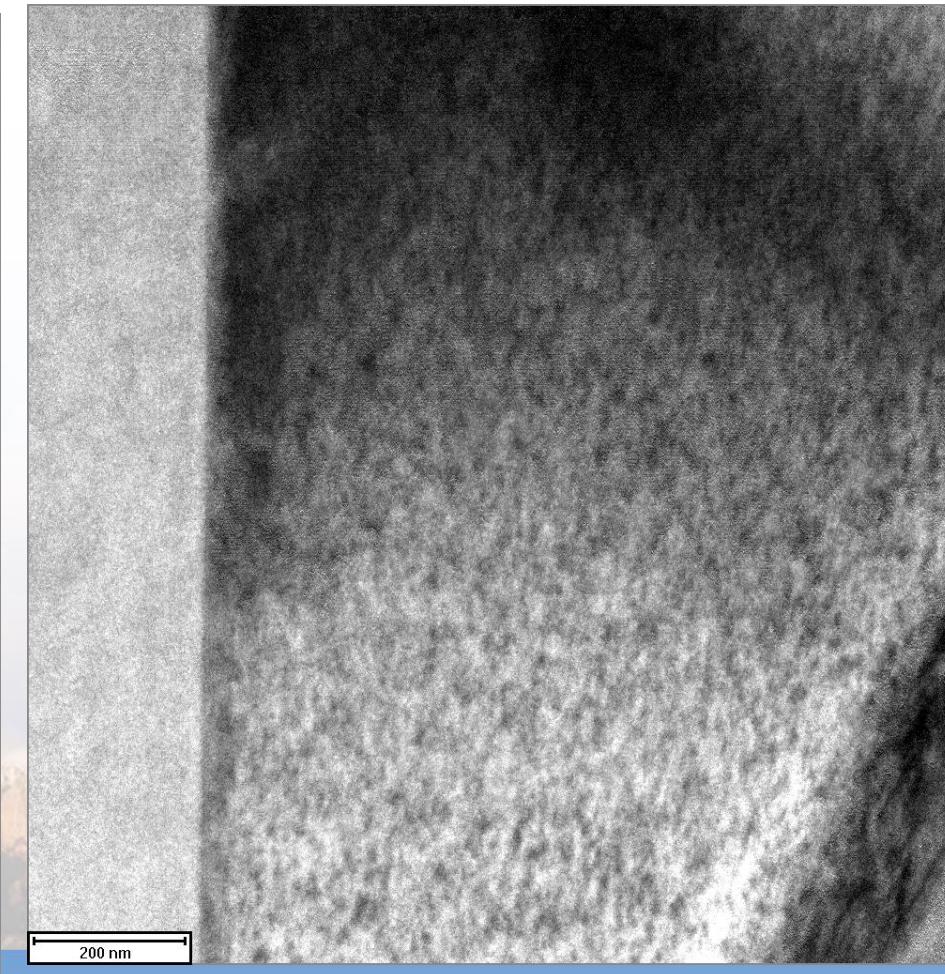
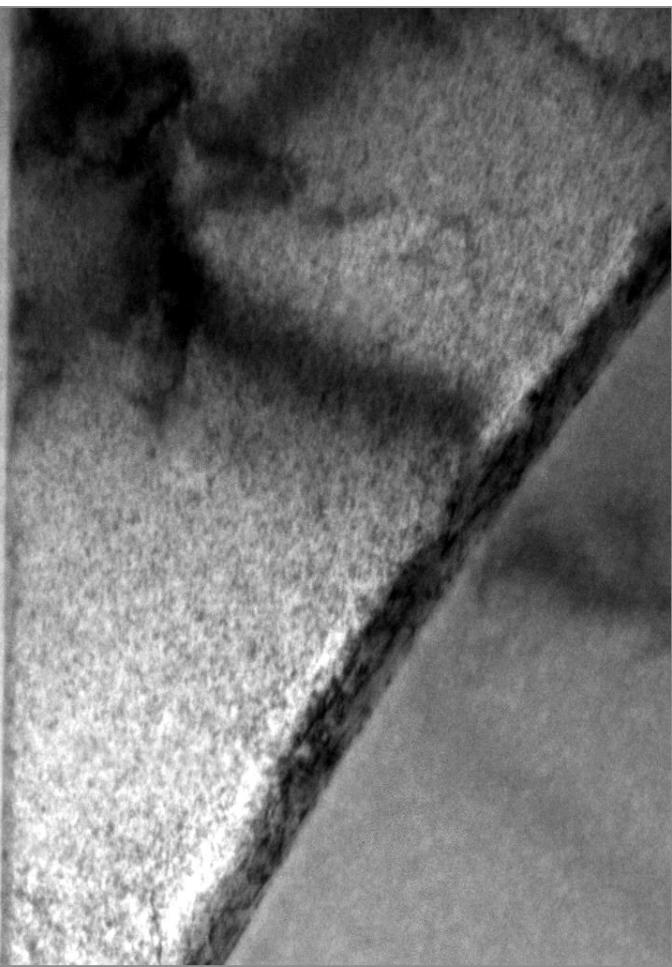
Ni³⁺ and D⁺ same time

before



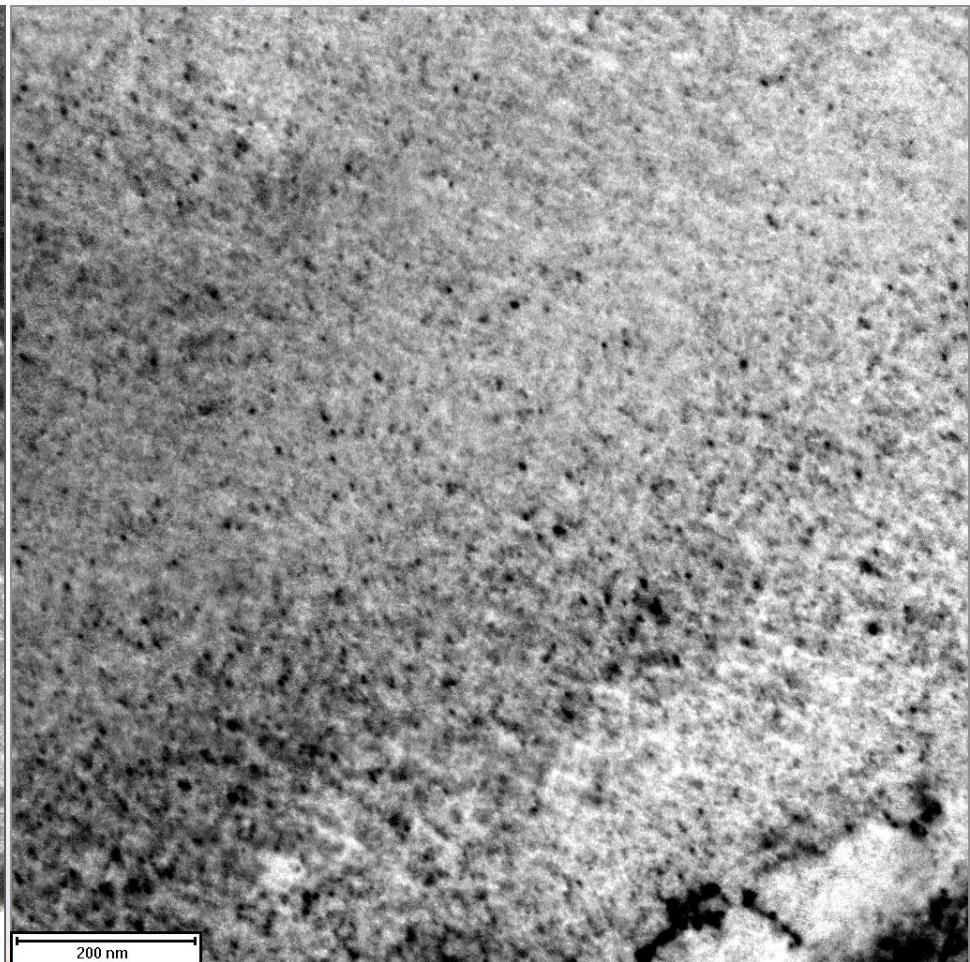
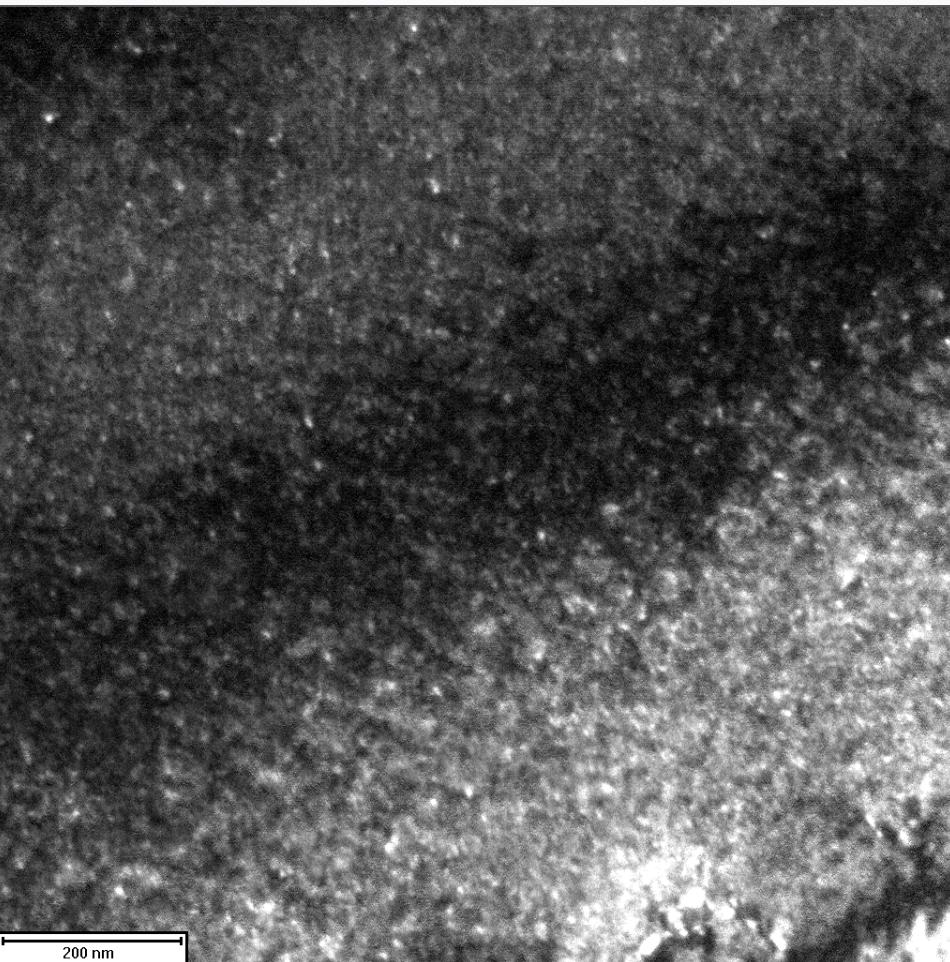


After Ni and D

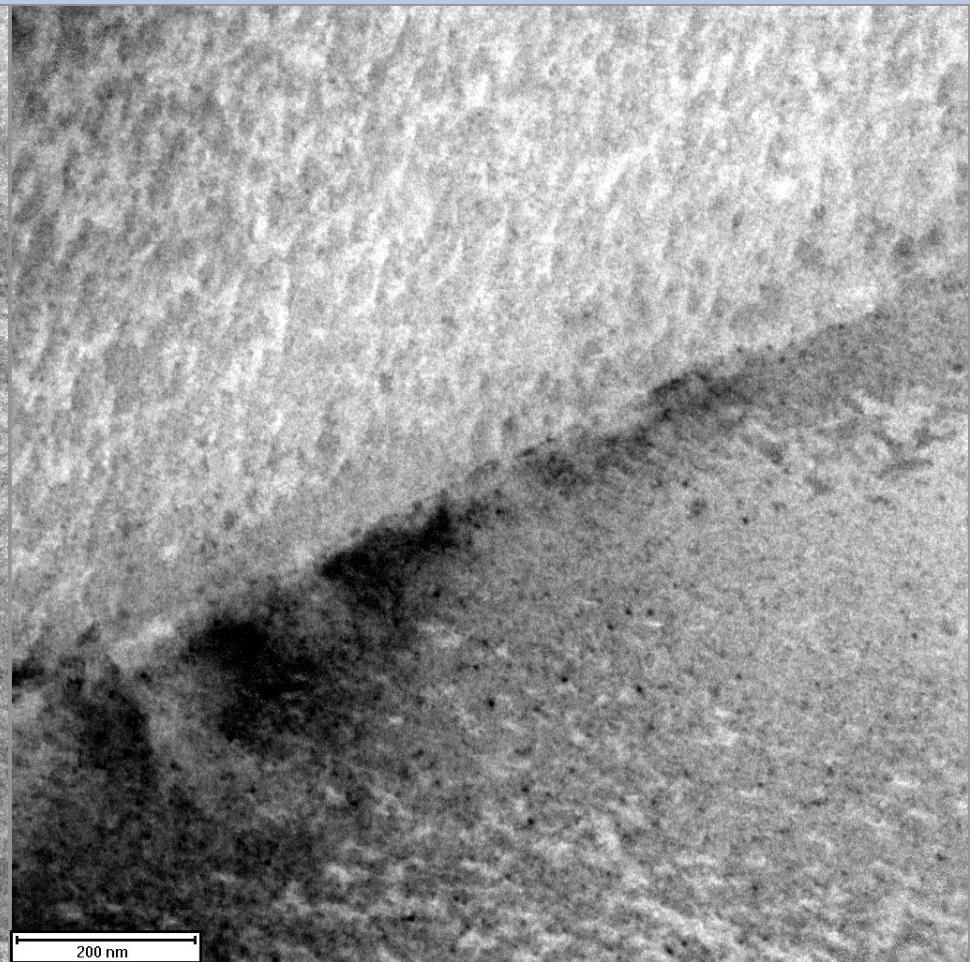
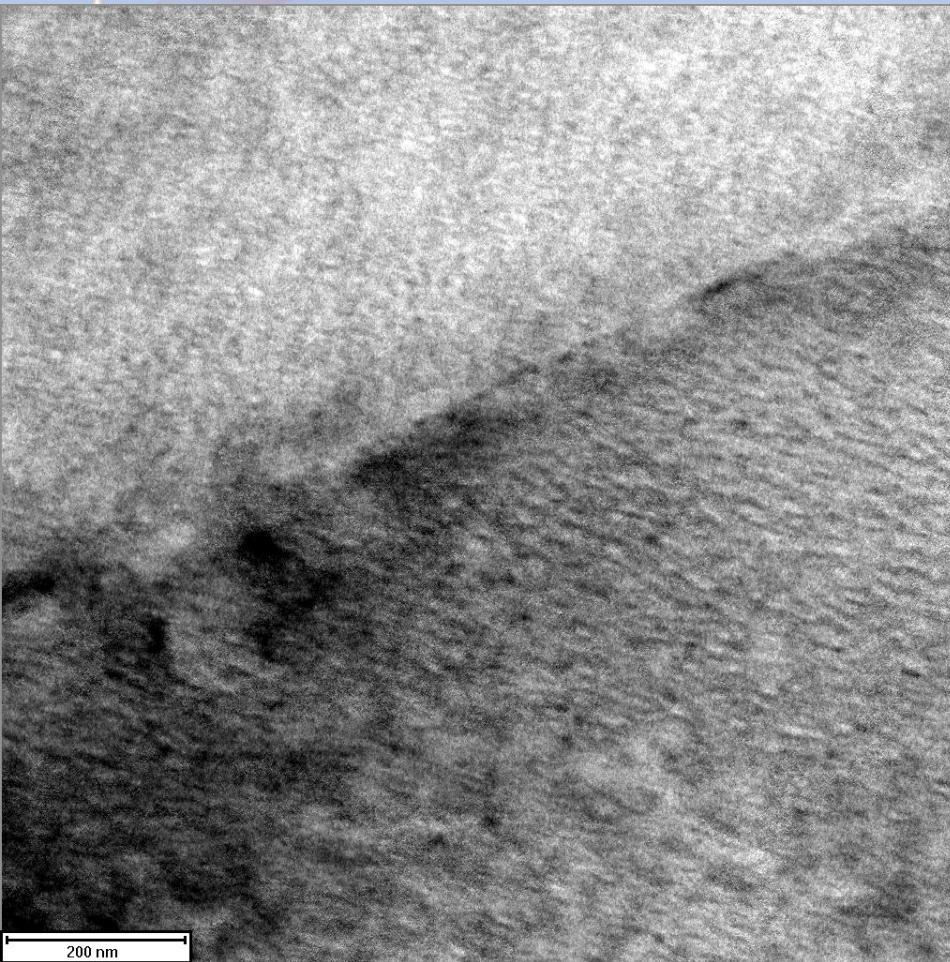


500 nm

200 nm



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Over and Under Focus

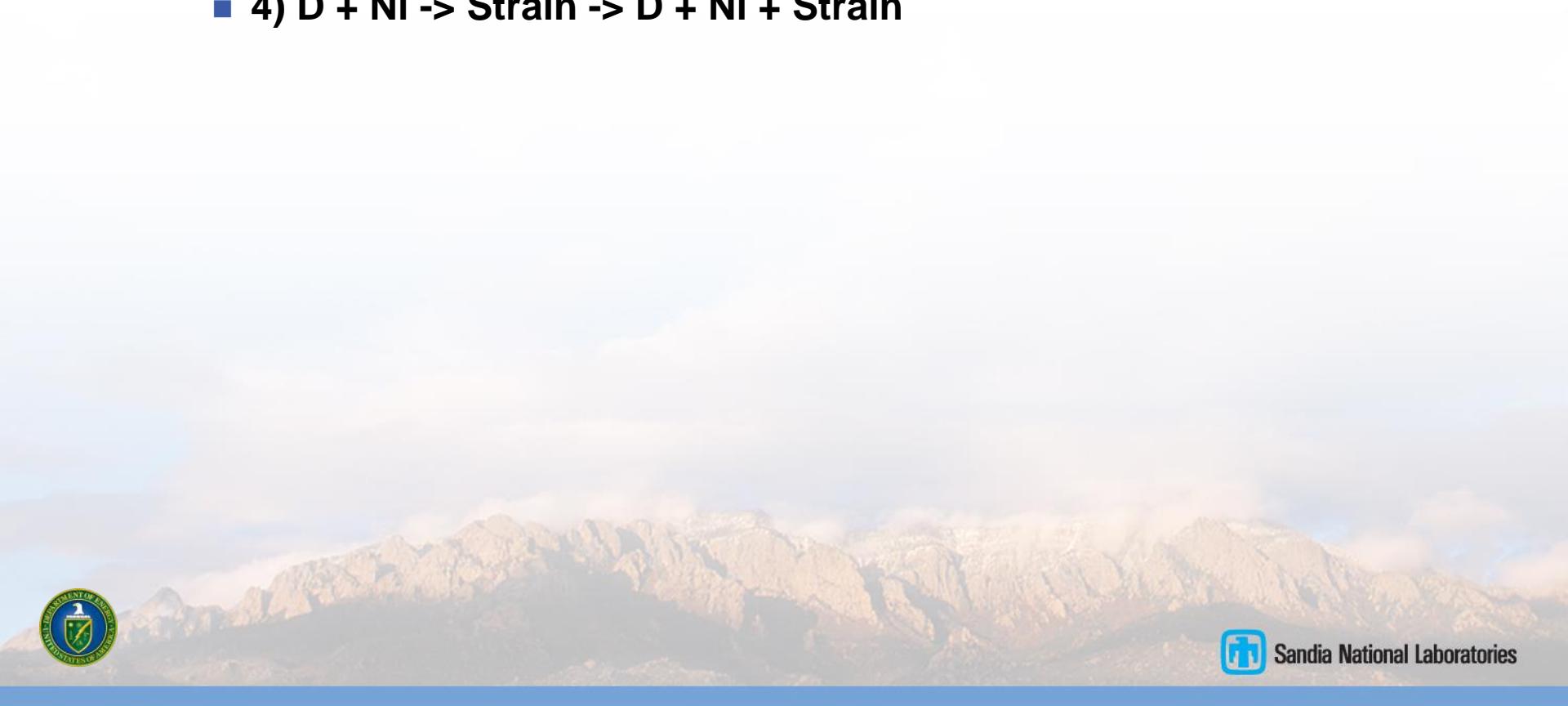


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Straining Plan

- 1) Control
- 2) D-> Strain-> D + Strain
- 3) Ni-> Strain -> Ni + Strain
- 4) D + Ni -> Strain -> D + Ni + Strain



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