

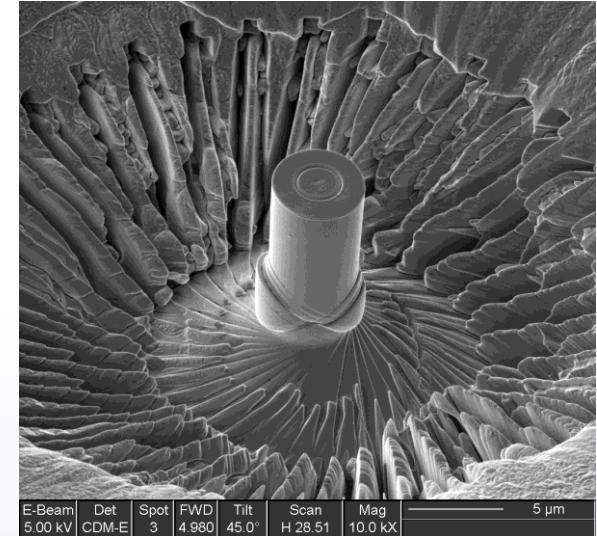
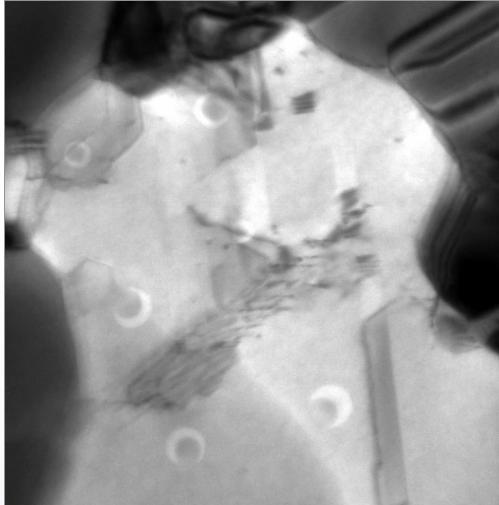
In situ Nanoscale Testing to Validate and Elucidate Mechanism for Predictive Modeling

SAND2014-17696PE

K. Hattar

Sandia National Laboratories

September 11, 2014



Collaborators:

- IBL: [D. Bufford](#), [D. Buller](#), [S. Pratt](#), [S. Rajasekhara](#), [J. Villone](#), and all the IBL staff
- Sandia: [T.E. Buchheit](#), [B. Boyce](#), [T.J. Boyle](#), [F.P. Doty](#), [P. Feng](#), [S. Goods](#), [B.A. Hernandez-Sanchez](#), [A.C. Kilgo](#), [P.G. Kotula](#), [J. Puskar](#), [M.J. Rye](#), [J.A. Scott](#), [P. Yang](#)
- External: [N. Li](#), [A. Misra](#), [L.N. Brewer](#), [S. Maloy](#), [A. McGinnis](#), [P. Rossi](#), [Protochips Inc.](#)



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What is Sandia National Laboratories?



96% of total NW parts



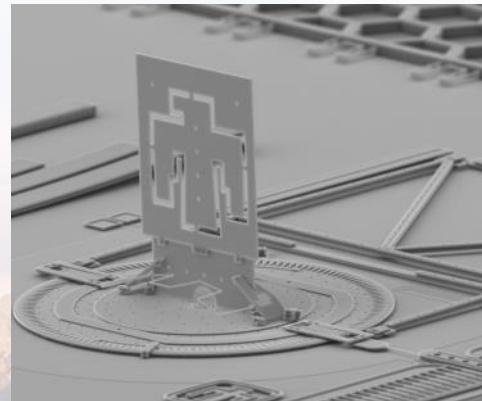
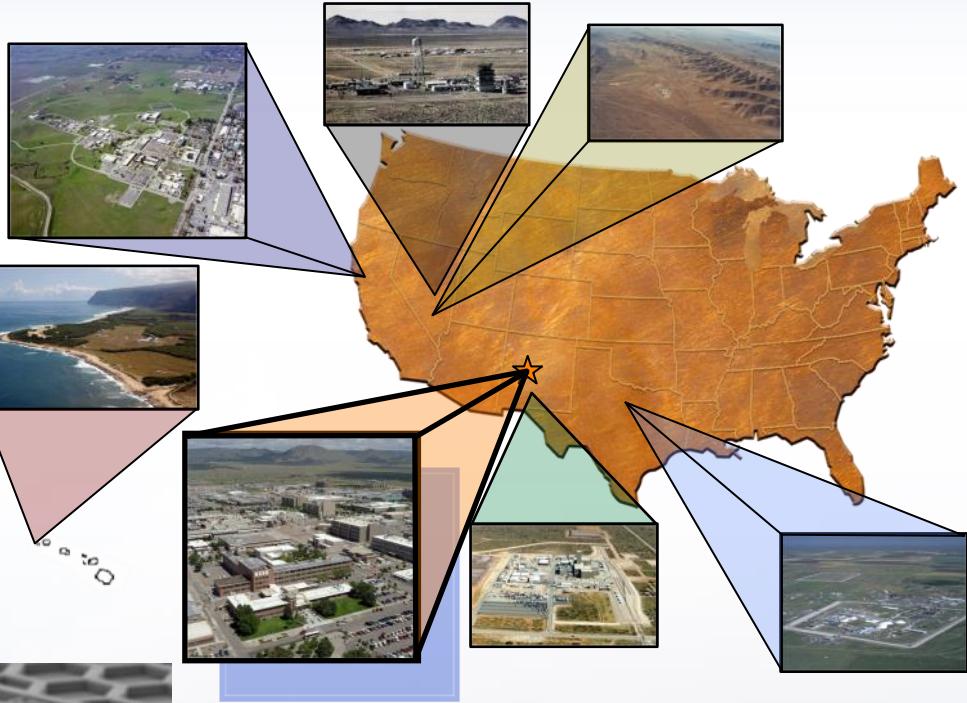
Sled track



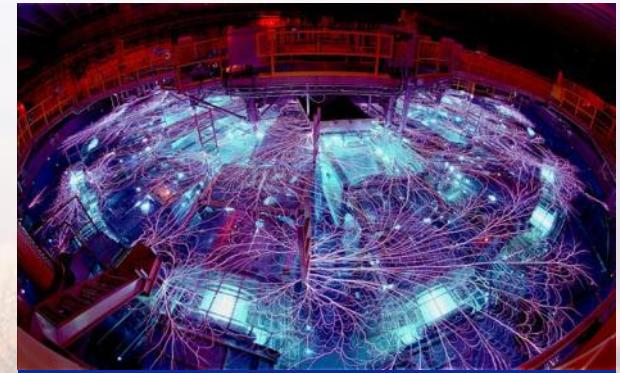
Renewable and alternative energy

4 Mission Areas

- Nuclear Weapons
- Defense Systems and Assessments
- Energy, Resources, and Nonproliferation
- Homeland Security and Defense

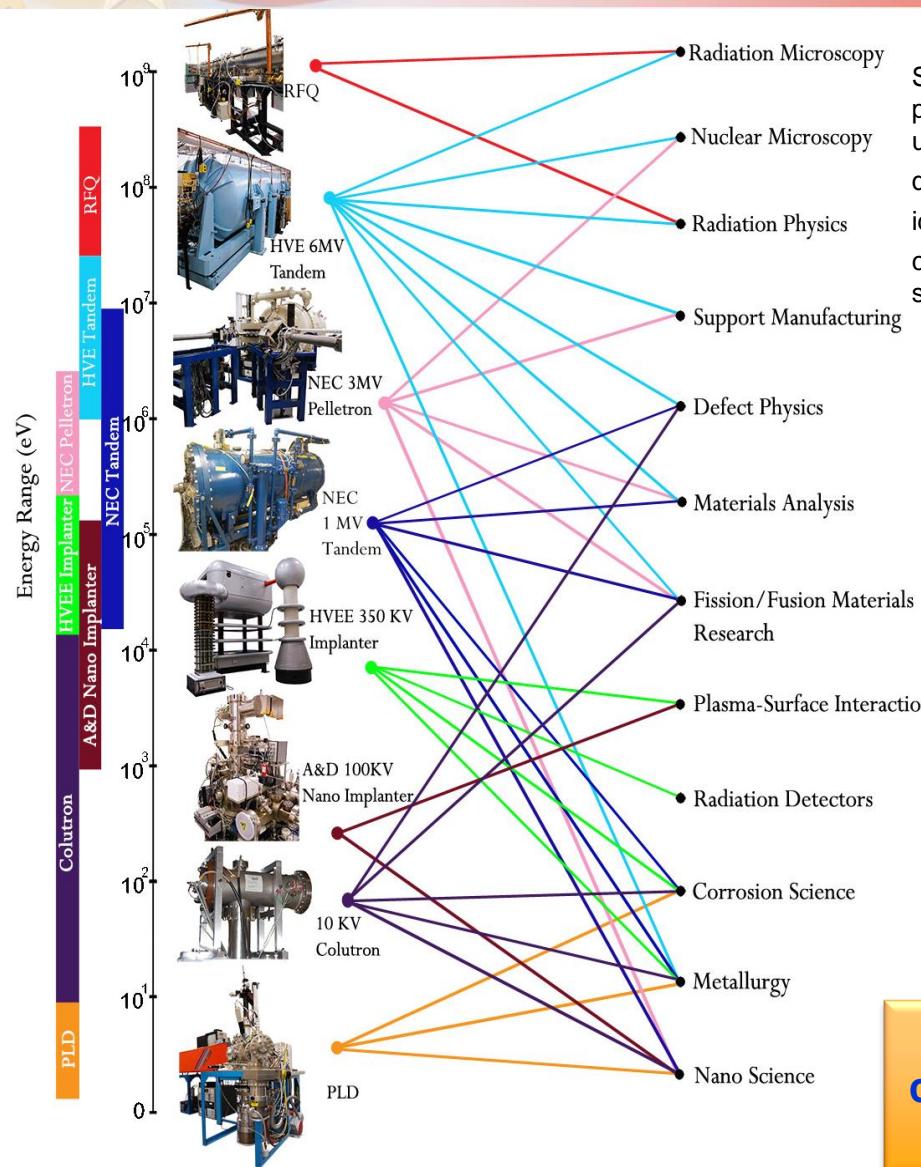


Clean room invented at SNL in 1963



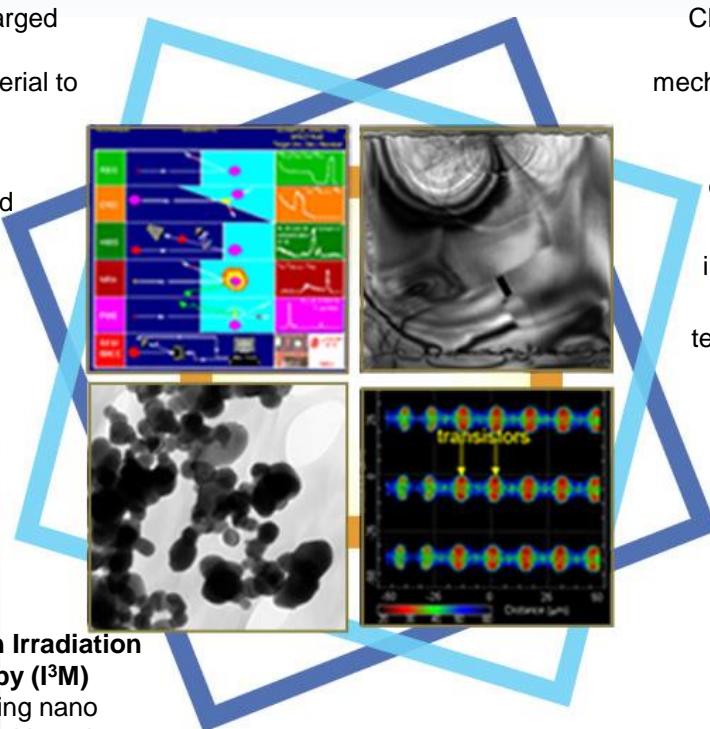
Z machine:
the world's most powerful X-ray source

Sandia's Ion Beam Laboratory



Ion Beam Analysis (IBA)

Shooting a charged particle at an unknown material to determine its identity, local chemistry, and structure.



Ion Beam Modification (IBM)

Changing the optical, mechanical, and chemical properties of materials via ion implantation to meet technological needs

In Situ Ion Irradiation Microscopy (I³M)

Bombarding nano samples with various particles and observing the changes in real time to understand how materials will behave in extreme environments.

The IBL has a unique and comprehensive capability ion beam set including and *In situ* Ion Irradiation Transmission Electron Microscopy.

Using ion emissions to determine the Radiation hardness of microelectronics, identifying potential weaknesses.

Potential Evolution of System Design

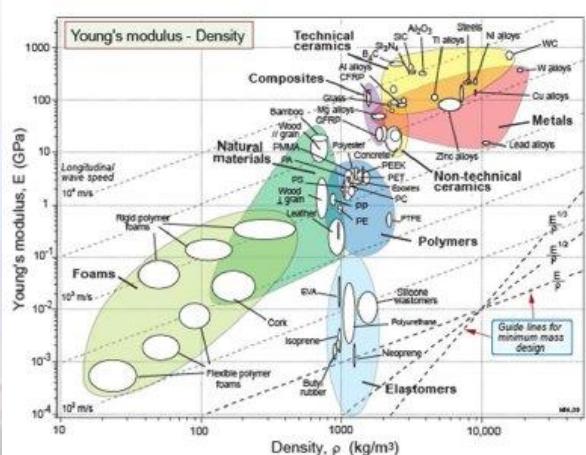
Use the Nearest Stone



to



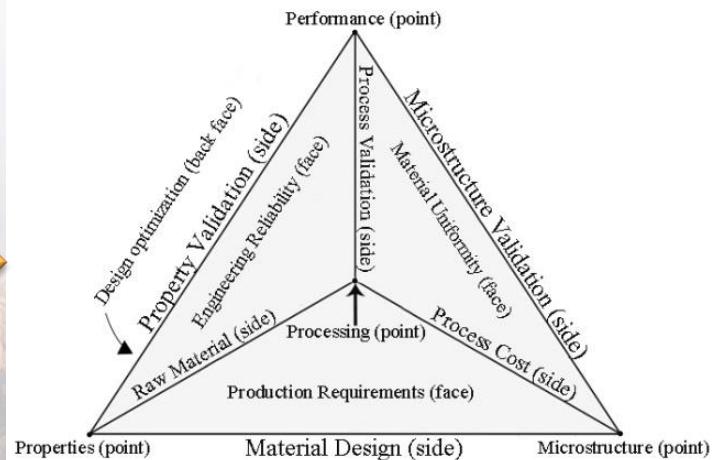
- Radar charts and Ashby plots of current material
- Accelerated and field testing
- **Scientist create a new materials. Engineers find an application**



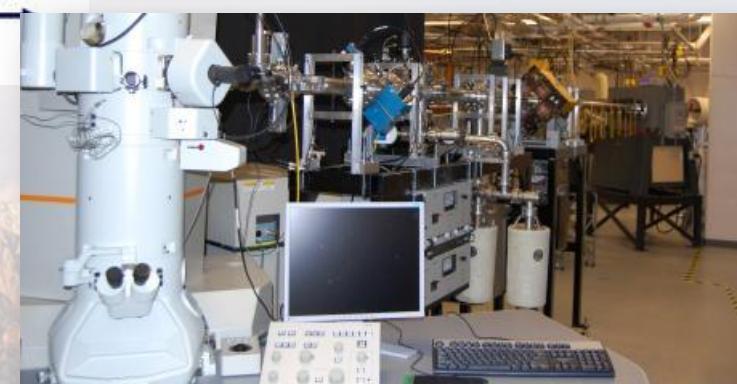
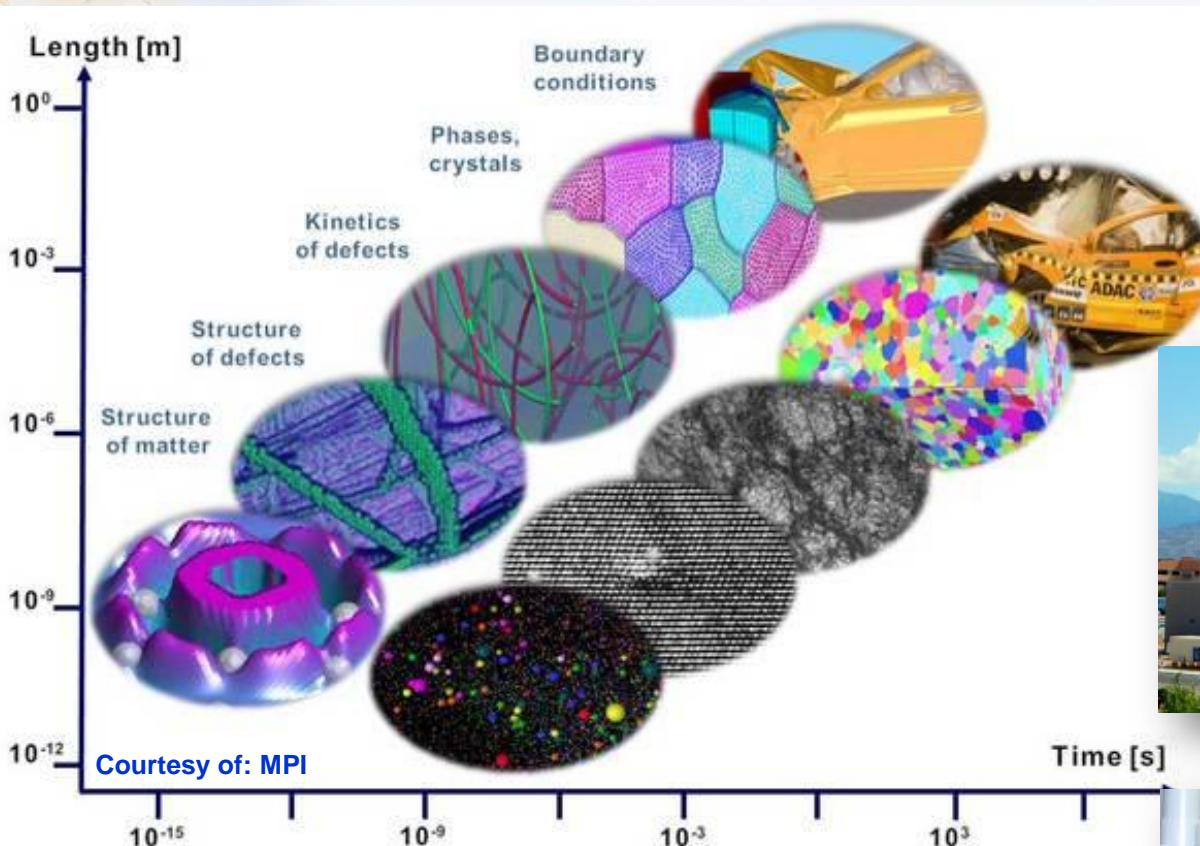
Materials by Design

- Physics-based approach
- Requires multiscale modeling
- **Engineers require given properties, Scientists tailor the chemistry and microstructure to achieve it.**

Great vision! We are making strides, but we are not there yet



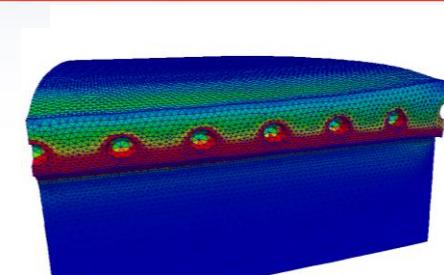
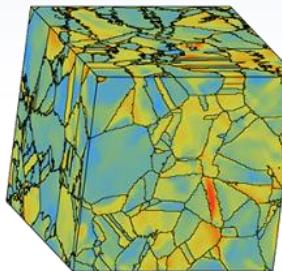
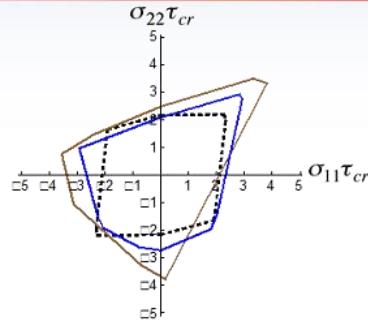
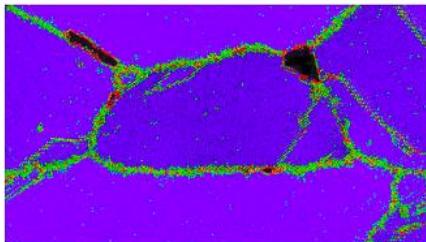
Investigating the nm Scale to Understand the km Scale to Understand Materials Response in the Extremes



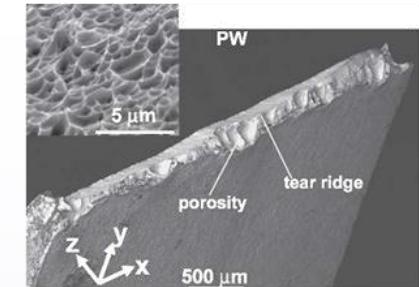
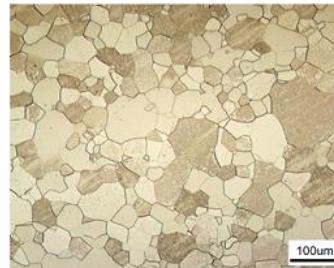
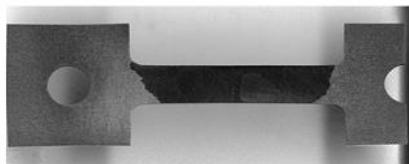
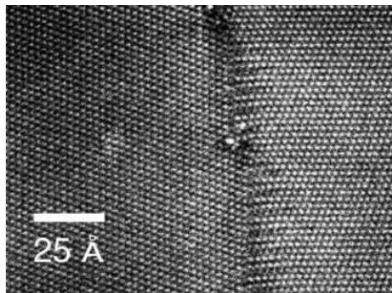
To develop predictive physics-based models, a fundamental understanding of the structure of matter, defects, and the kinetics of structural evolution in the environments of interest are needed

Approach: Multiscale simulation & experiments are needed to understand and predict the sources of material variability

simulations



experiments



Atomic scale phenomena
 $10^{-9} \text{ m } 10^{-9} \text{ s}$

Single crystal behavior
 $10^{-6} \text{ m } 10^0 \text{ s}$

Microstructural effects
 $10^{-3} \text{ m } 10^3 \text{ s}$

Material performance
 $10^0 \text{ m } 10^6 \text{ s}$

Atoms-up: Develop physics-based models to provide scientific insight

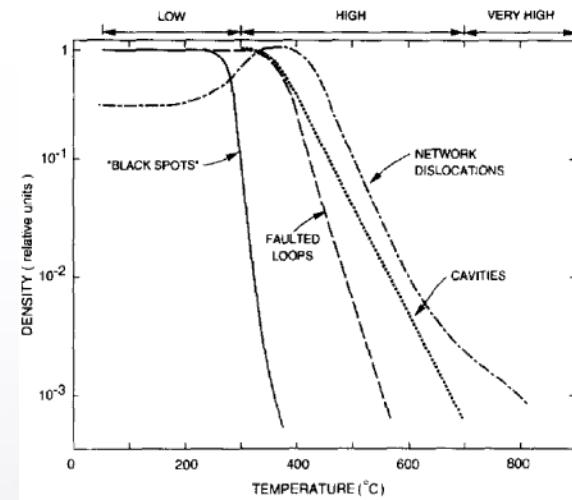
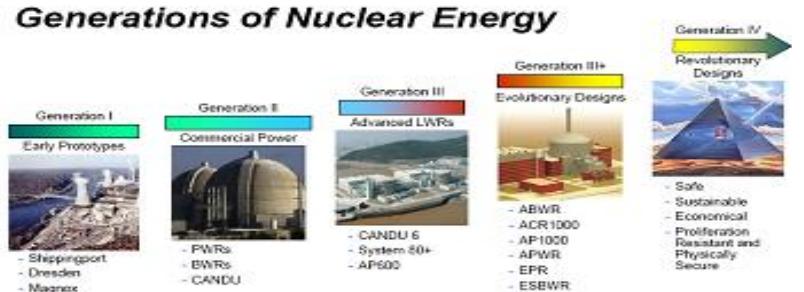
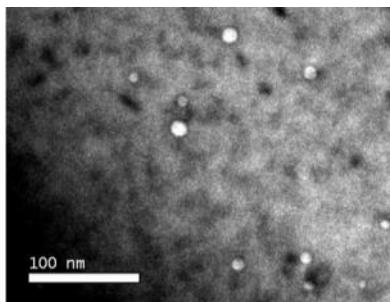
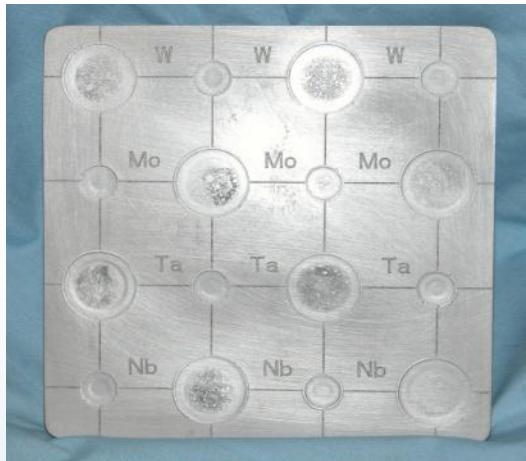
Continuum-down: Augment engineering-scale models to provide improved fidelity



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Sandia's Approach to Rapid Material Validation for Advanced Materials Necessary for New Reactors

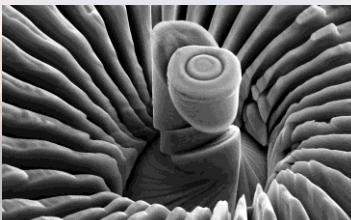
- Advanced materials are needed
- Several theories exist for the desired microstructure
- New materials have been made
- Current neutron fluxes require decades for testing



Local Composition (Material Design) +
Local Microstructural Control (Ion Irradiation)

Microstructural Characterization (XTEM)

Mechanical Properties (small-scale testing)



Validating Comparison to Neutron Irradiation Experiments + Investigation into new materials



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Testing of Irradiated Stainless Steels

Collaborators: L.N. Brewer, T.E. Buchheit and A.J. Kilgo

- Micropillar is difficult for many polycrystalline materials
 - Due to the dependence of FIB milling rate on orientation

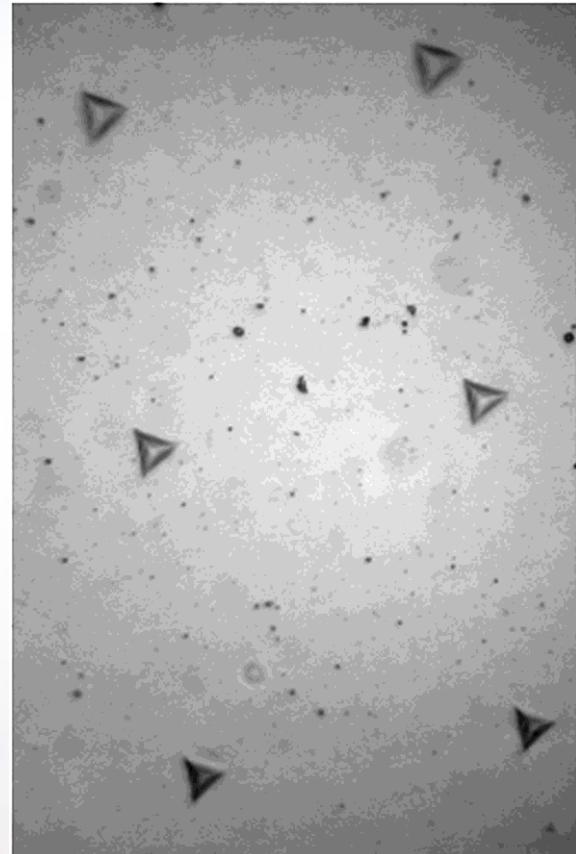
To validate the approach:

1. Metals previously tested by Neutron Irradiation must be tested
2. The effect of temperature and various ion characteristics must be considered

Thus, we irradiated

- 420, 409, and 316L SS
- Approximately 10 dpa, 40 dpa, and 100 dpa
- Temperatures of 400 °C, 500 °C, and 600 °C

**Three steel compositions were irradiated under various conditions.
Nanoindentation was selected as the optimal small scale testing method.**

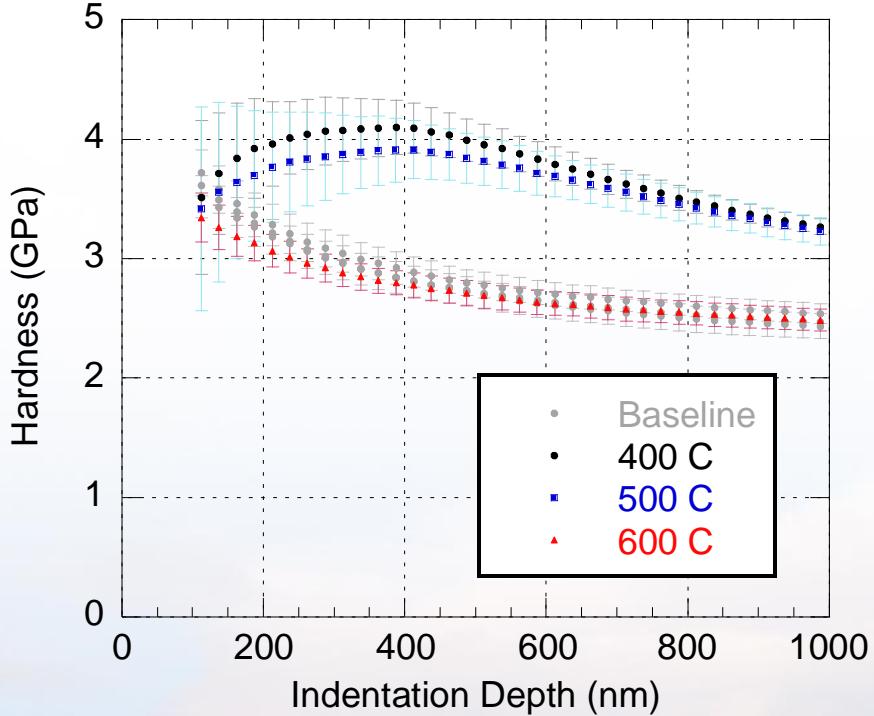


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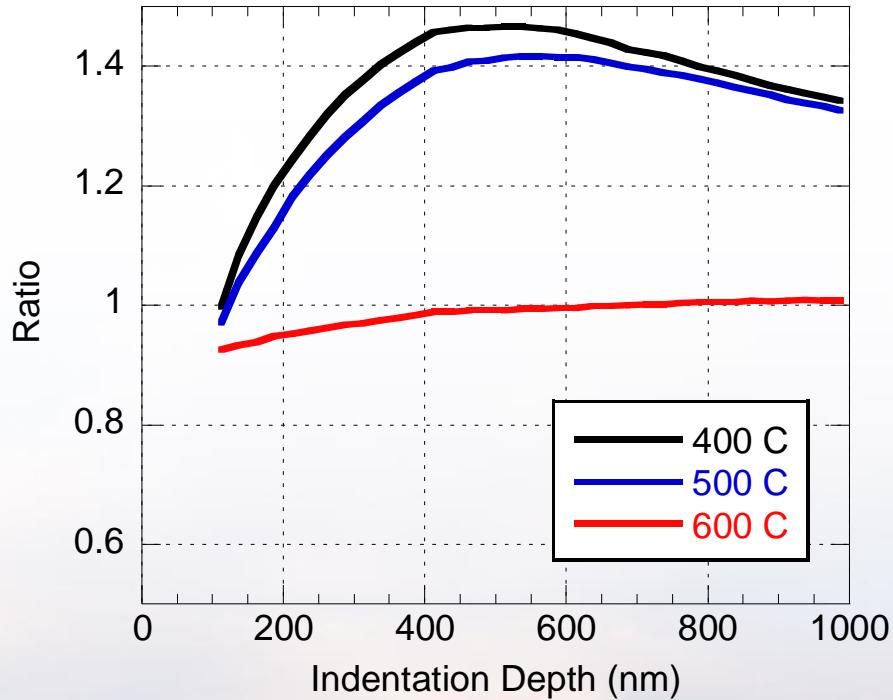
Berkovitch Indentation of 100 dpa Irradiated Samples

Collaborators: L.N. Brewer, T.E. Buchheit and A.J. Kilgo

Hardness vs. Indentation Depth
Comparison of 100 dpa measurements



Baseline to Implanted Region Hardness Ratio
vs. Indentation Depth - 100 dpa experiments



At 100 dpa, the hardness difference between 400 °C and 500 °C sample and the control microstructure has increased.



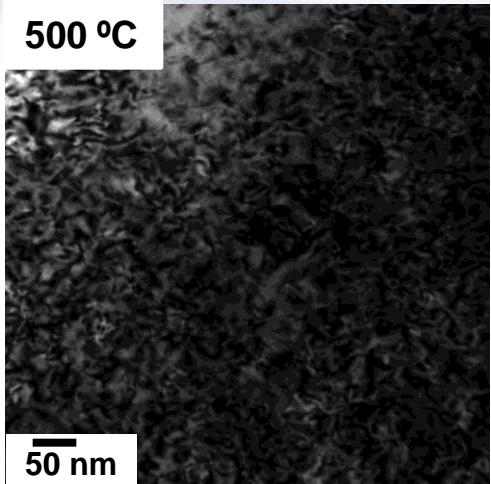
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Microstructural Evolution between 500 °C and 600 °C

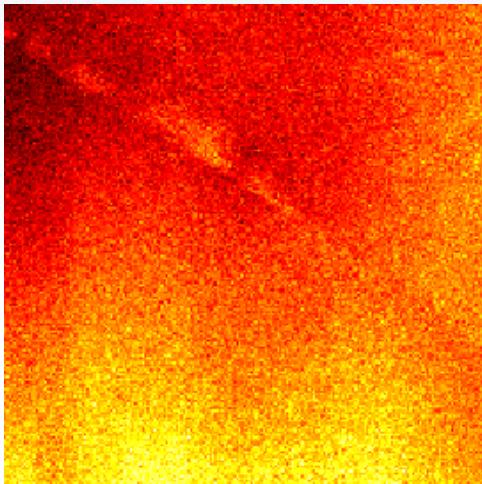
316L Stainless Steel: 100dpa, 20 MeV Nickel Ions

Collaborators: L.N. Brewer, A.J. Kilgo, P. Kotula

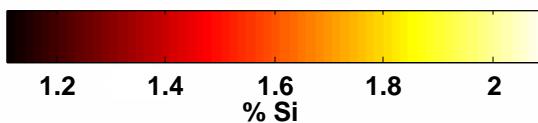
500 °C



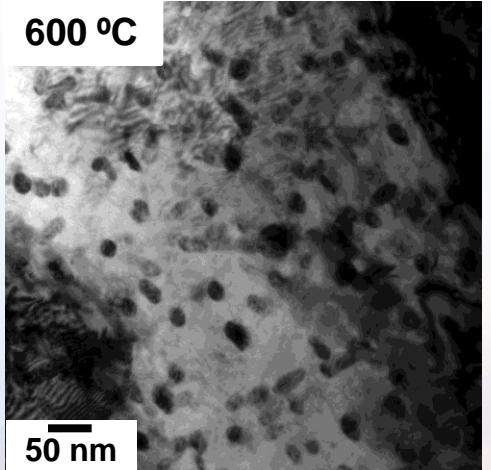
50 nm



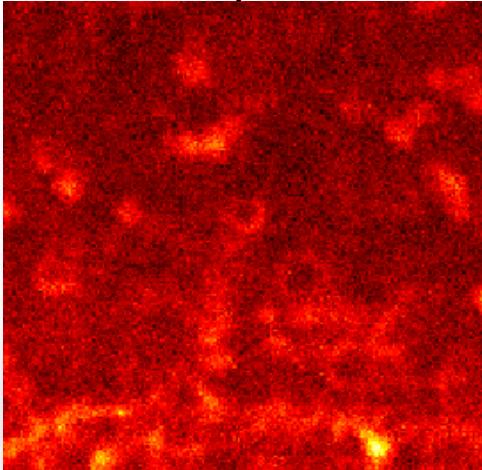
- Large number of small defects present in the irradiated region
- No significant segregation of either the Ni or Si constituents



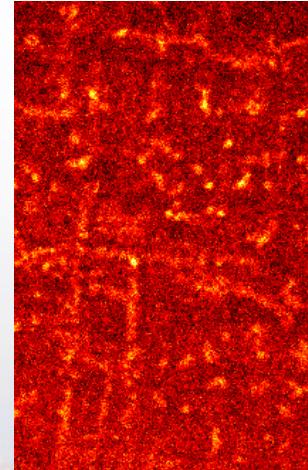
600 °C



50 nm

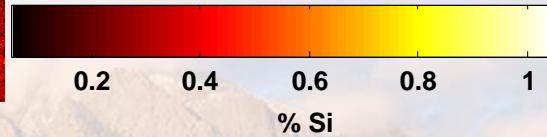


500 nm



1 μm x 2 μm

- Voids are formed and are self-ordered
- Significant segregation of either the Ni or Si constituents



Ni and Si rich regions appear to self-organize and sometimes surround voids at 600 °C, but not 500 °C



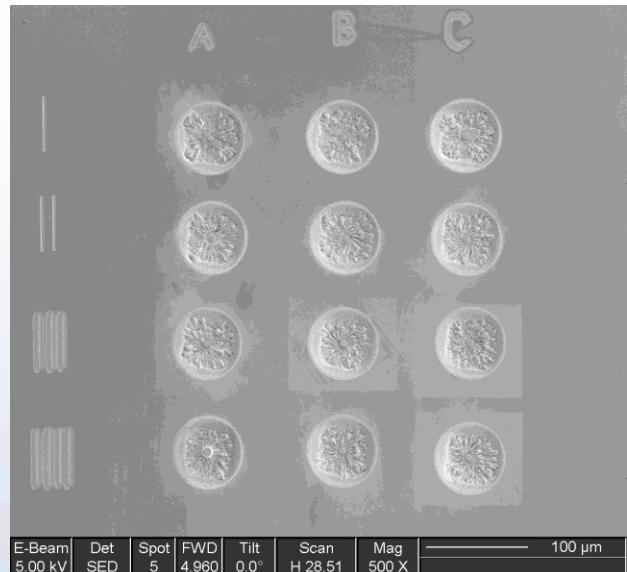
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Micropillar Compression Experiments

Collaborators: M.J. Rye, L.N. Brewer, B. Boyce

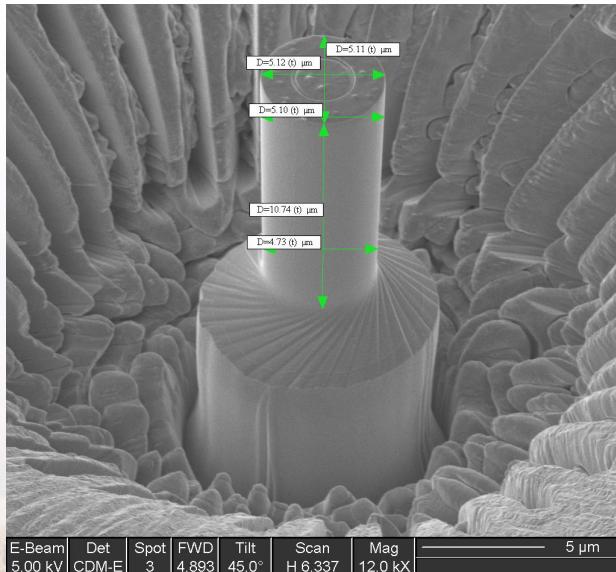
Sample Preparation:

- Copper single crystals (FCC)
- Different crystallographic orientations: (100), (110), and (111)
- Self-ion Implants at 30 MeV to 0 (control), 50 dpa, and 100 dpa.



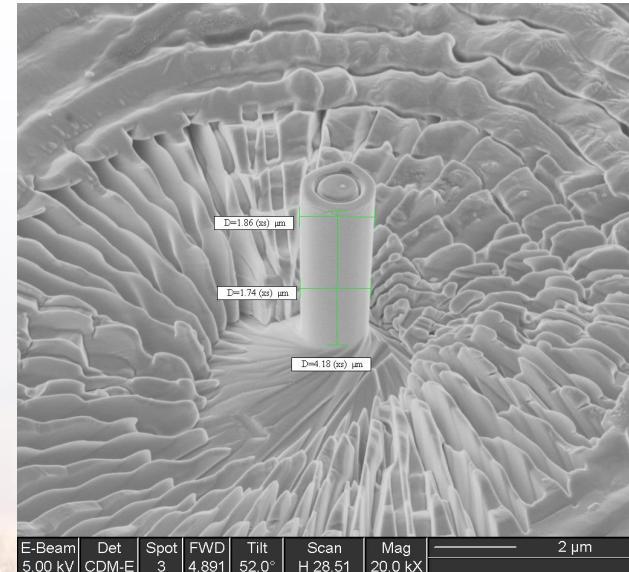
Pillar Manufacturing:

- We employed Uchic's FIB lathe machining process for straight-walled cylinders.
- Array of at least 9 nominally identical pillars tested per condition to assess statistical variability.
- Height varies from 4 μm to 10 μm



Compression Testing:

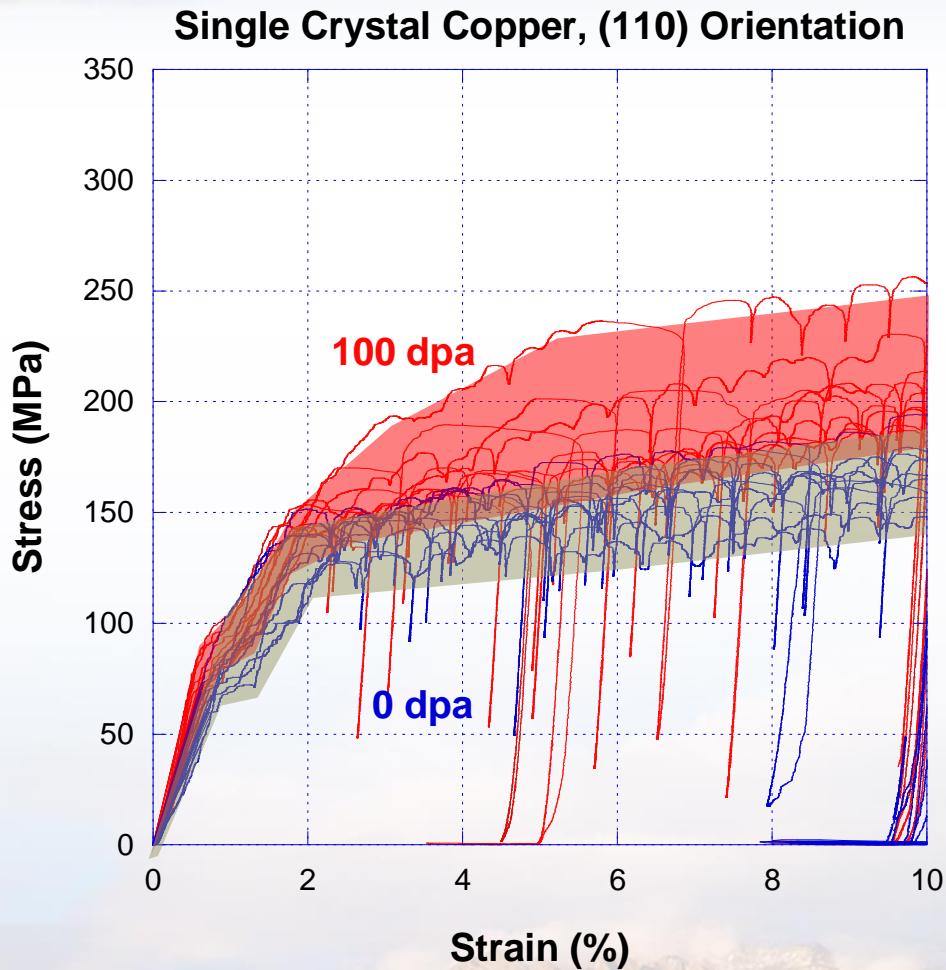
- Hysitron Performech Nanoindenter permits <1 nm and <1 μN resolution.
- 25 μm flat ended cone indenter in feedback displacement control, rather than typical force control.
- Pillars compressed 10% strain at a strain rate of 0.025 s^{-1} .



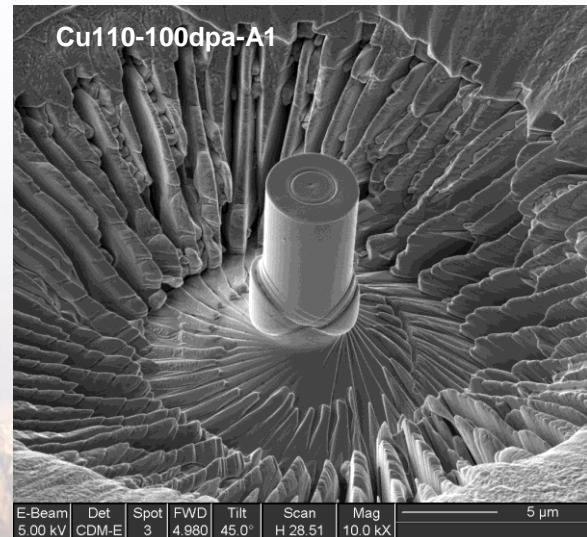
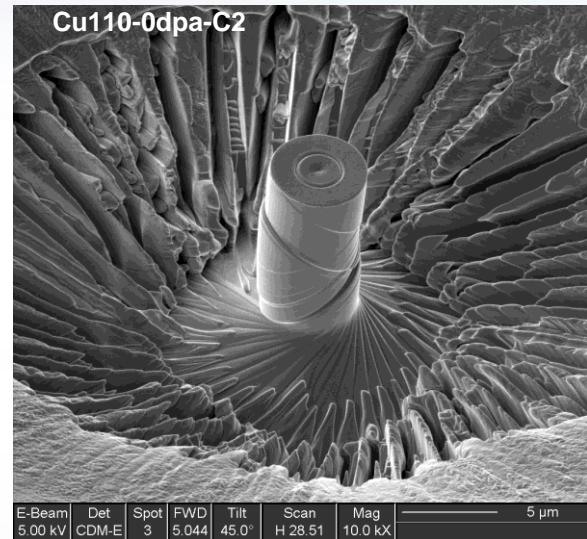
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Large Micropillar Compression

Collaborators: M.J. Rye, L.N. Brewer, B. Boyce



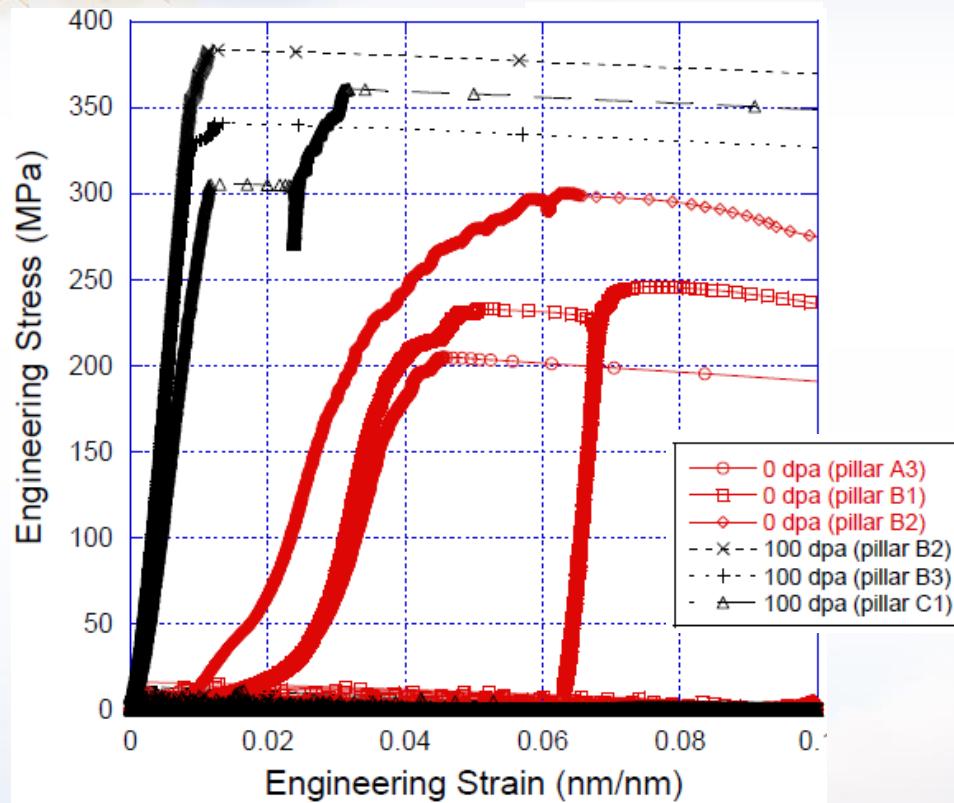
Minimal difference between the control and irradiated 10 μm -tall pillars. Slip occurred in the bottom fraction of the pillars.



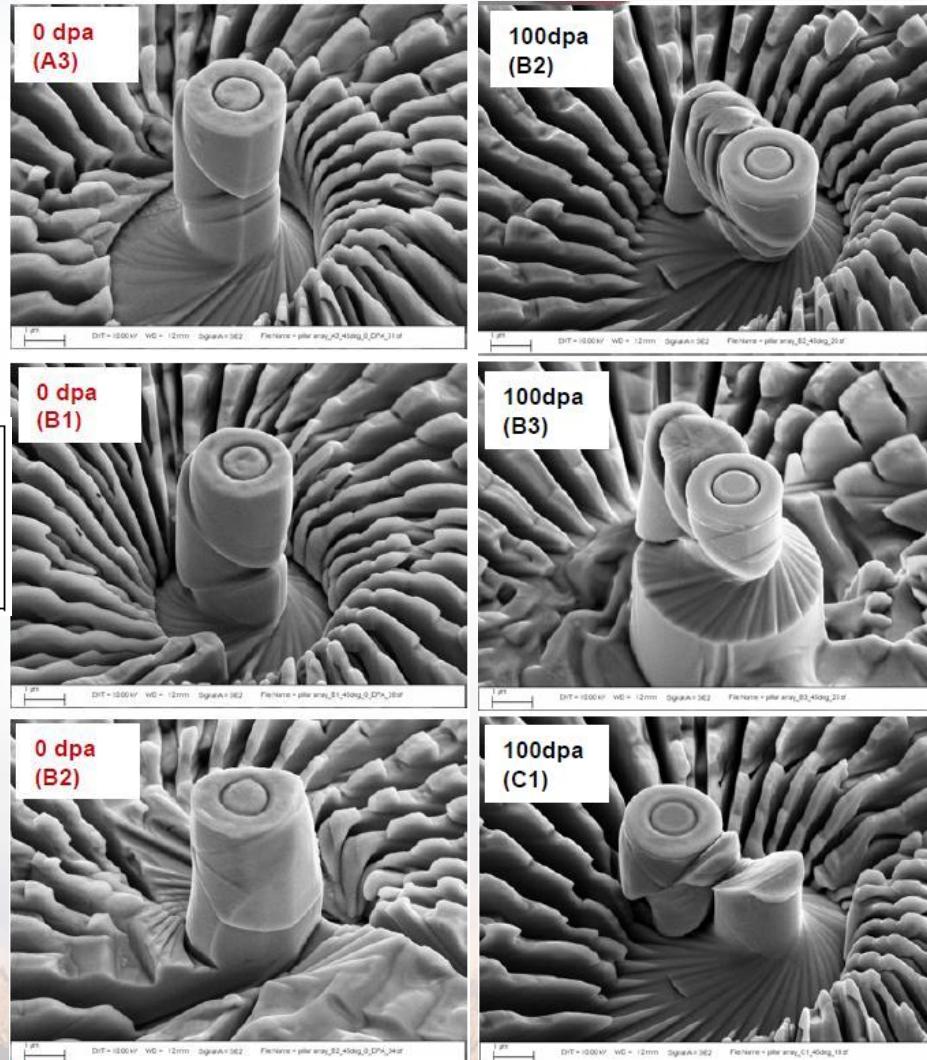
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Intermediate Micropillar Compression

Collaborators: M.J. Rye, L.N. Brewer, B. Boyce



5 μ m-tall pillars show greater distinction with catastrophic failure



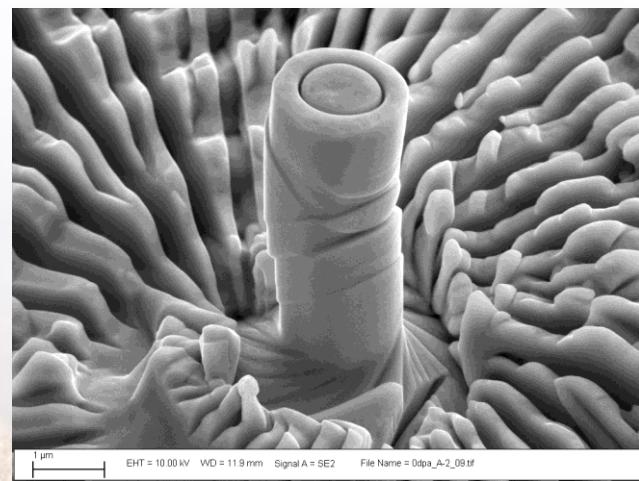
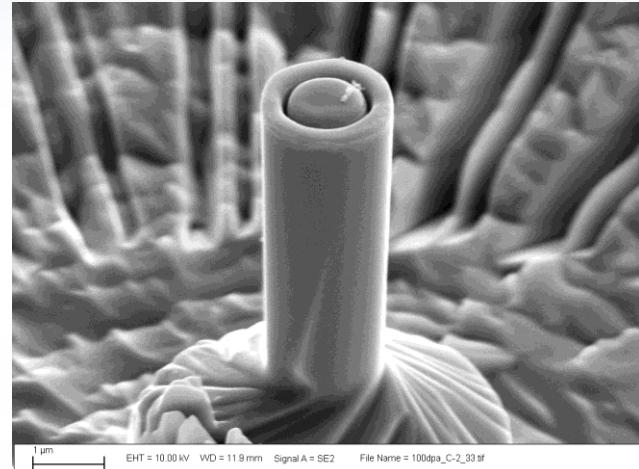
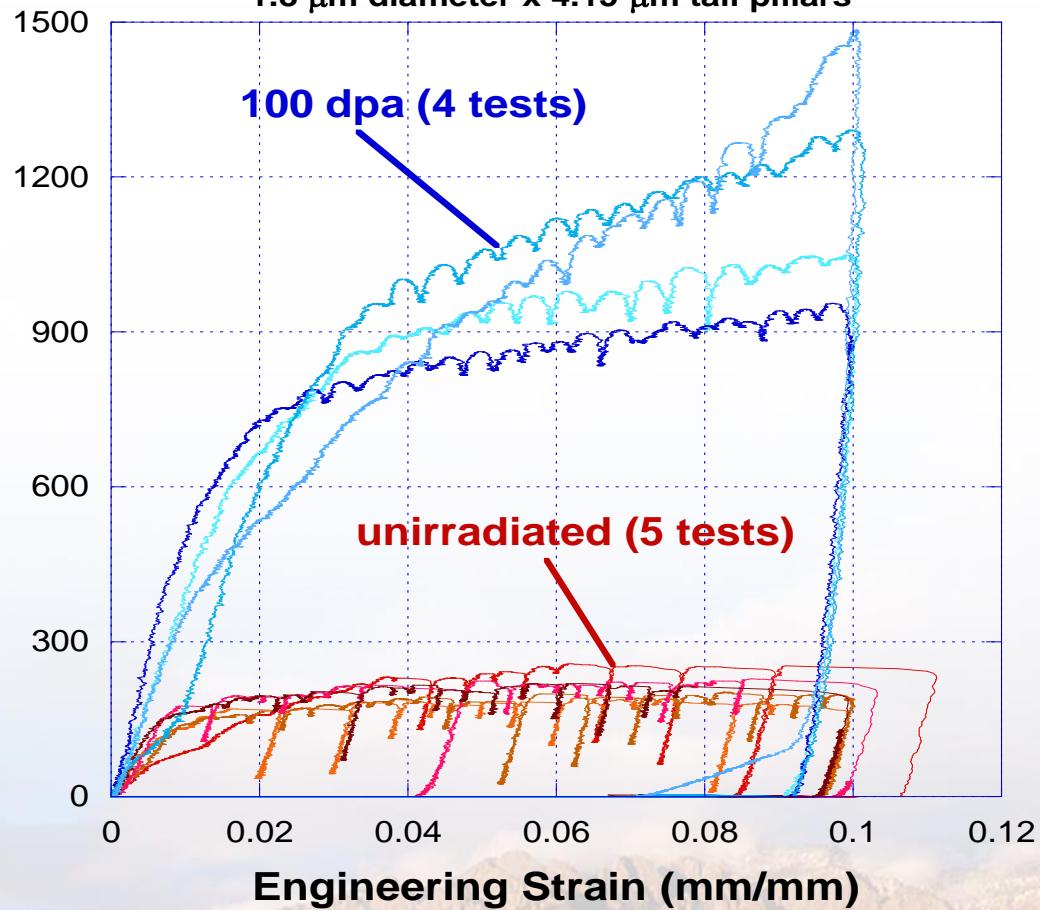
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Small Micropillar Compression

Collaborators: M.J. Rye, L.N. Brewer, B. Boyce

Single Crystal Cu - (110) orientation
1.8 μm diameter x 4.15 μm tall pillars

Engineering Stress (MPa)



Initial tests indicate that the 4 μm -tall pillars are 5 times stronger
and show no signs of slip band formation

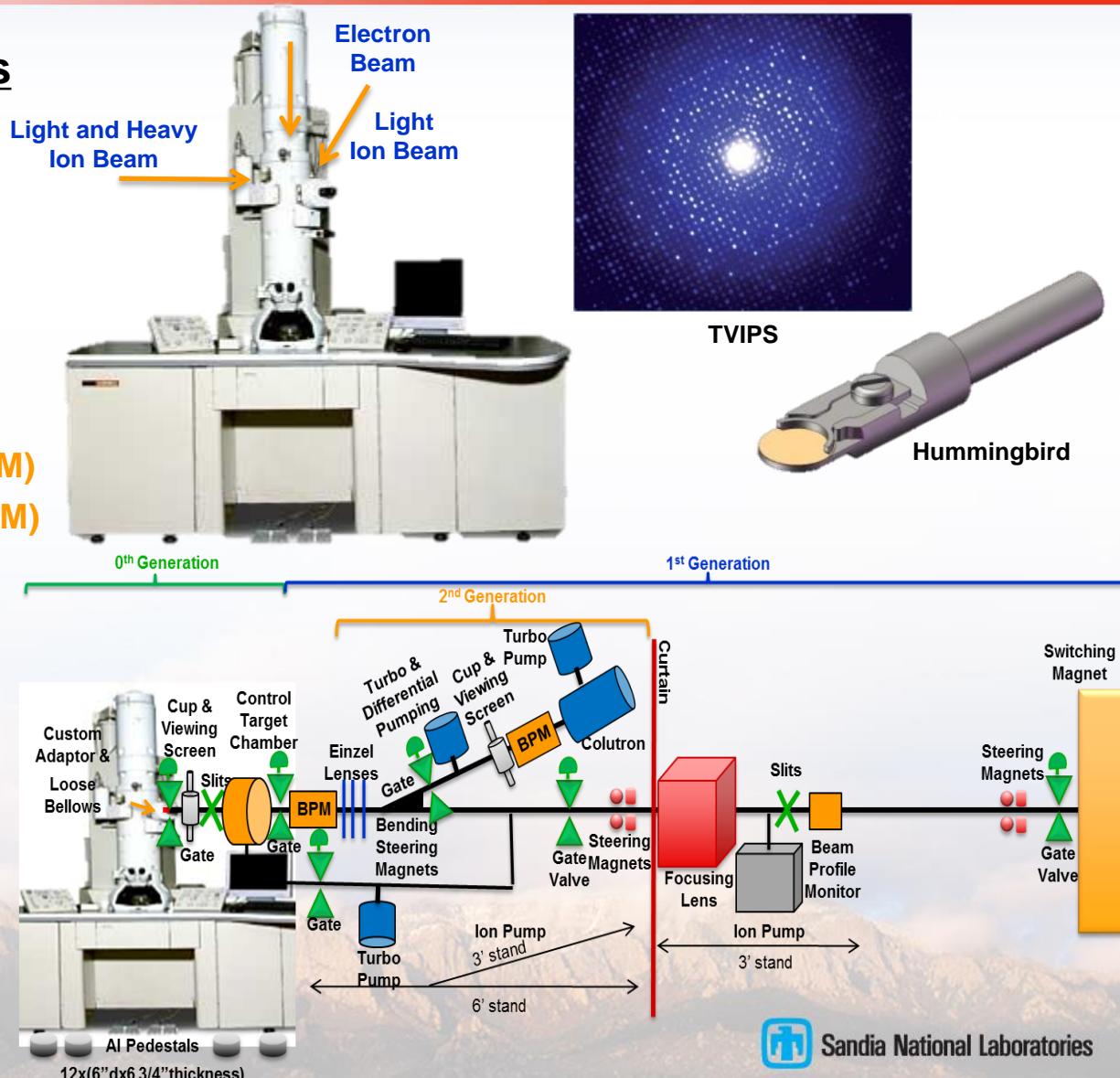


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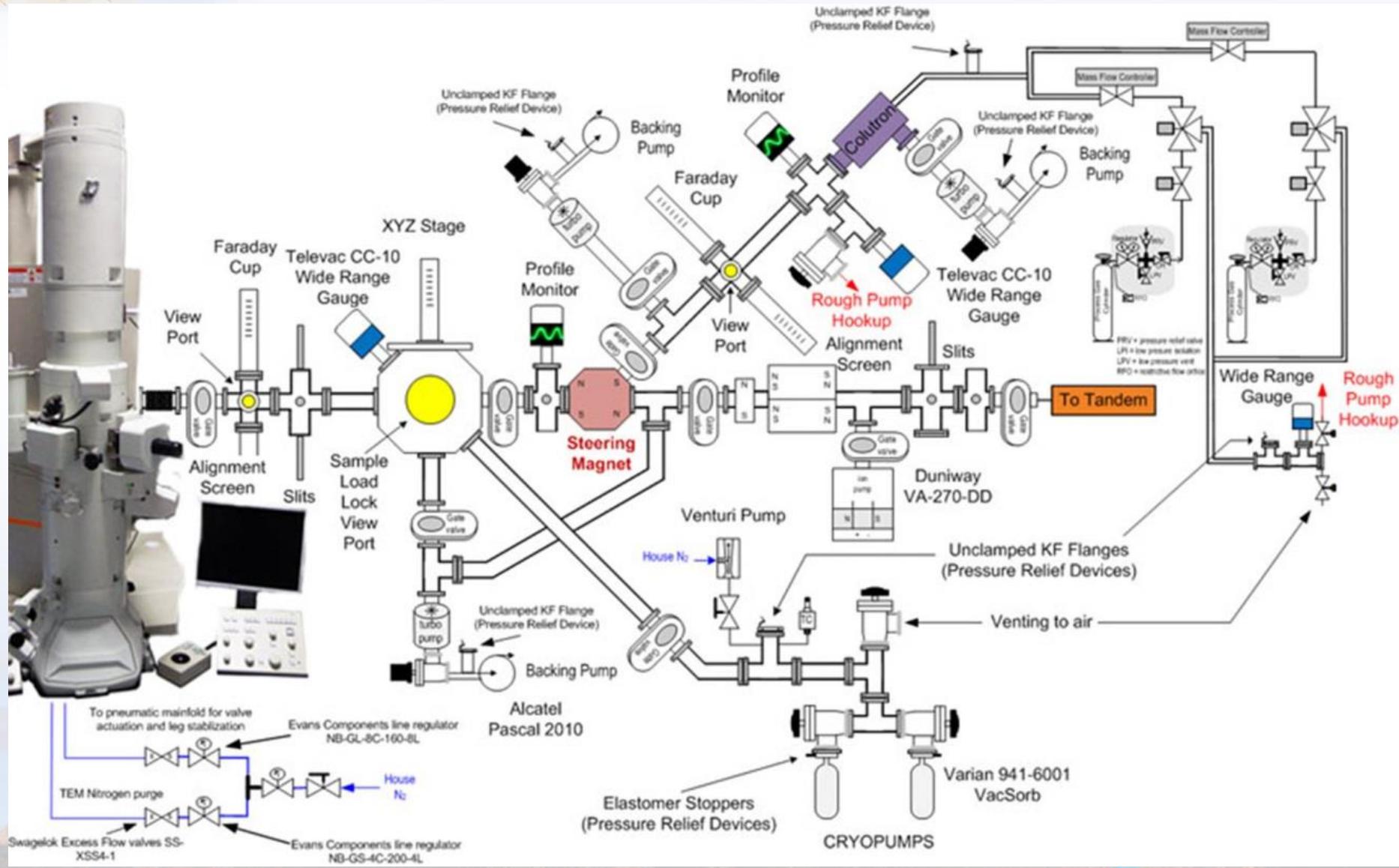
In situ Ion Irradiation TEM Facility

Proposed Capabilities

- 200 kV LaB₆ TEM
- Ion beams considered:
 - Range of Sputtered Ions
 - 10 keV D²⁺
 - 10 keV He⁺
- All beams hit same location
- Nanosecond time resolution (DTEM)
- Procession scanning (EBSD in TEM)
- *In situ* PL, CL, and IBIL
- *In situ* vapor phase stage
- *In situ* liquid mixing stage
- *In situ* heating
- Tomography stage (2x)
- *In situ* cooling stage
- *In situ* electrical bias stage
- *In situ* straining stage



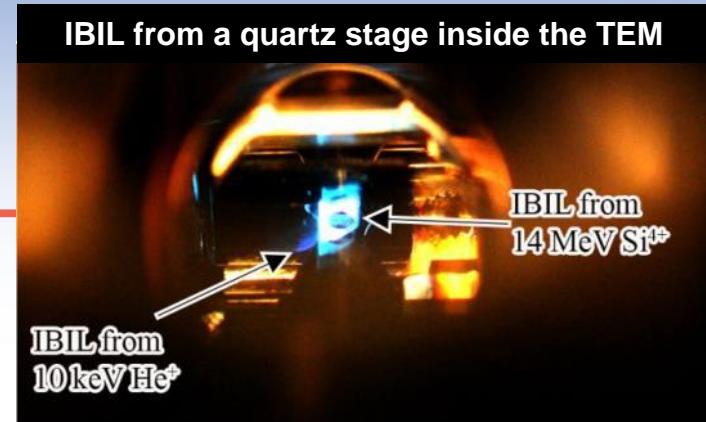
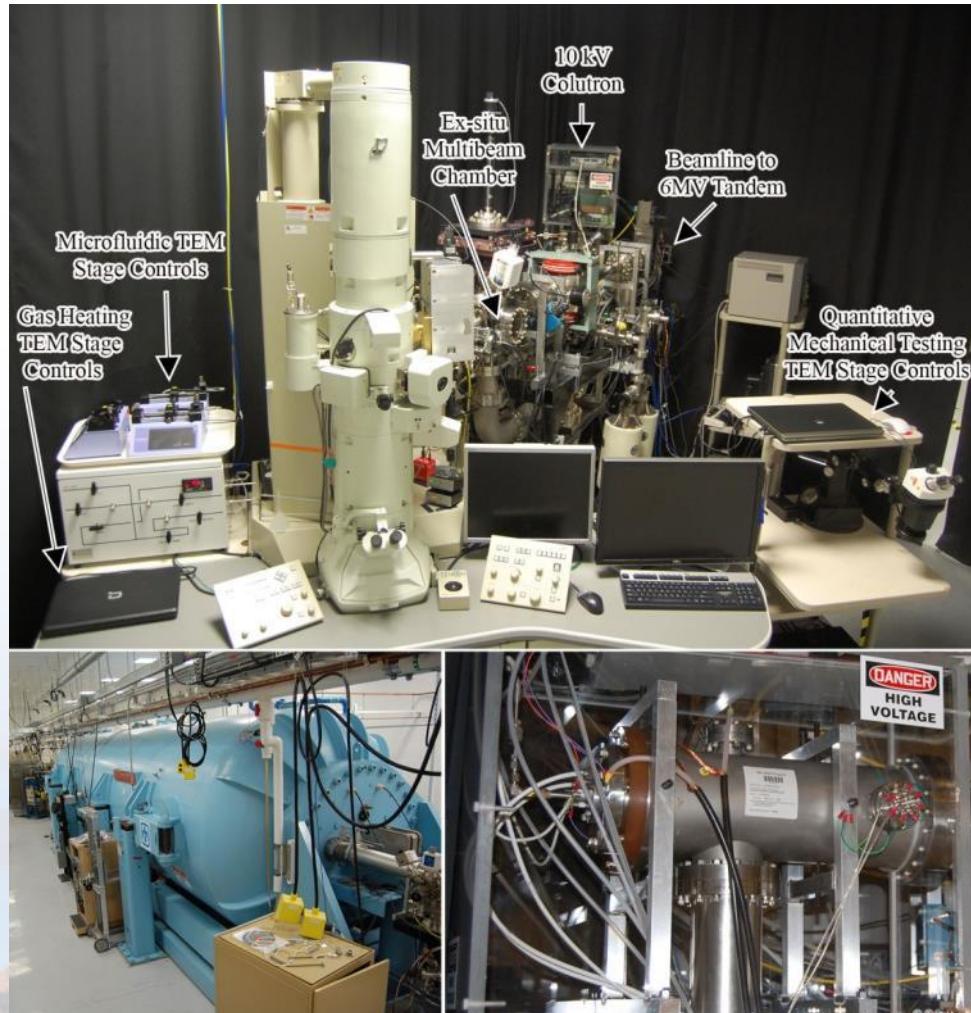
Schematic of the *In situ* TEM Beamline



Sandia's Concurrent *In situ* Ion Irradiation TEM Facility

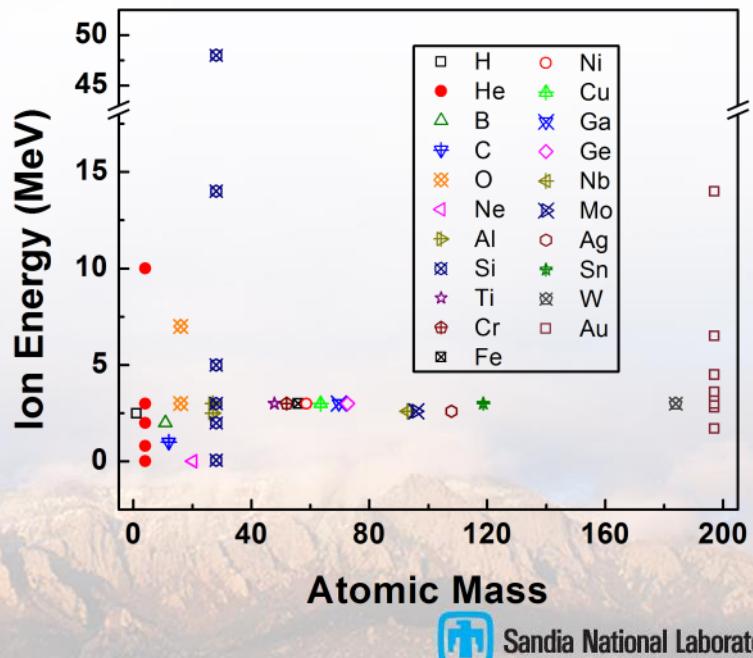
Collaborator: D.L. Buller

10 kV Colutron - 200 kV TEM - 6 MV Tandem



Direct real time observation
of ion irradiation,
ion implantation, or both
with nanometer resolution

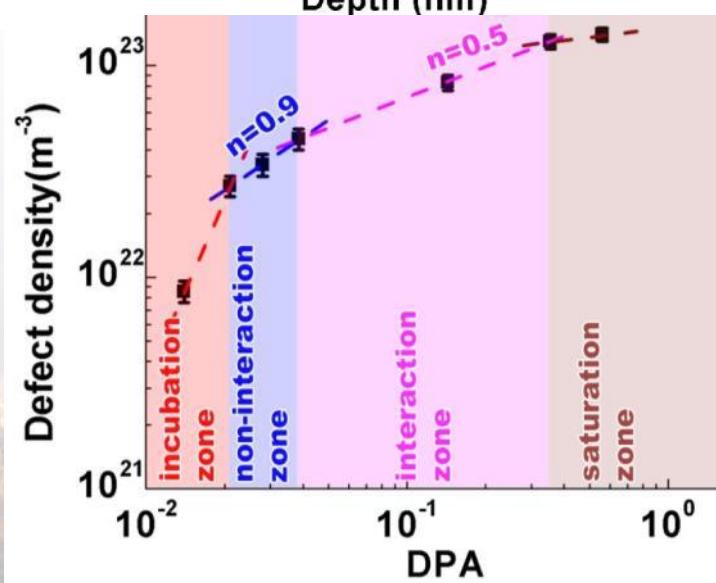
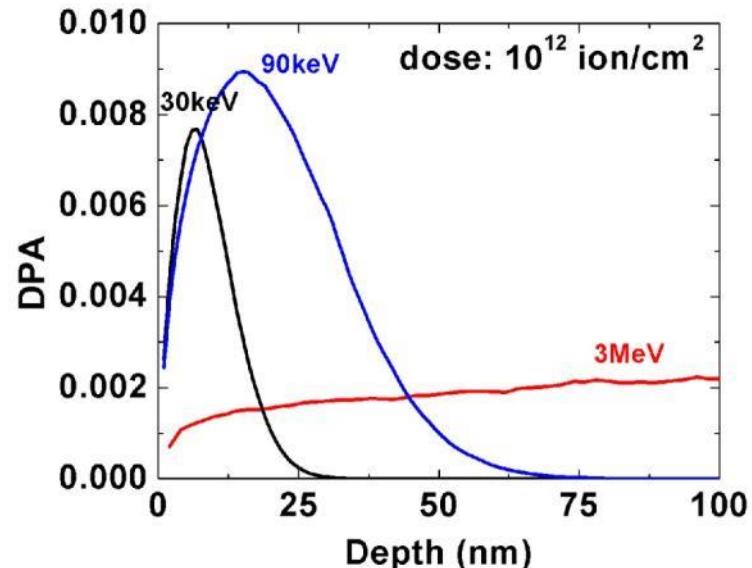
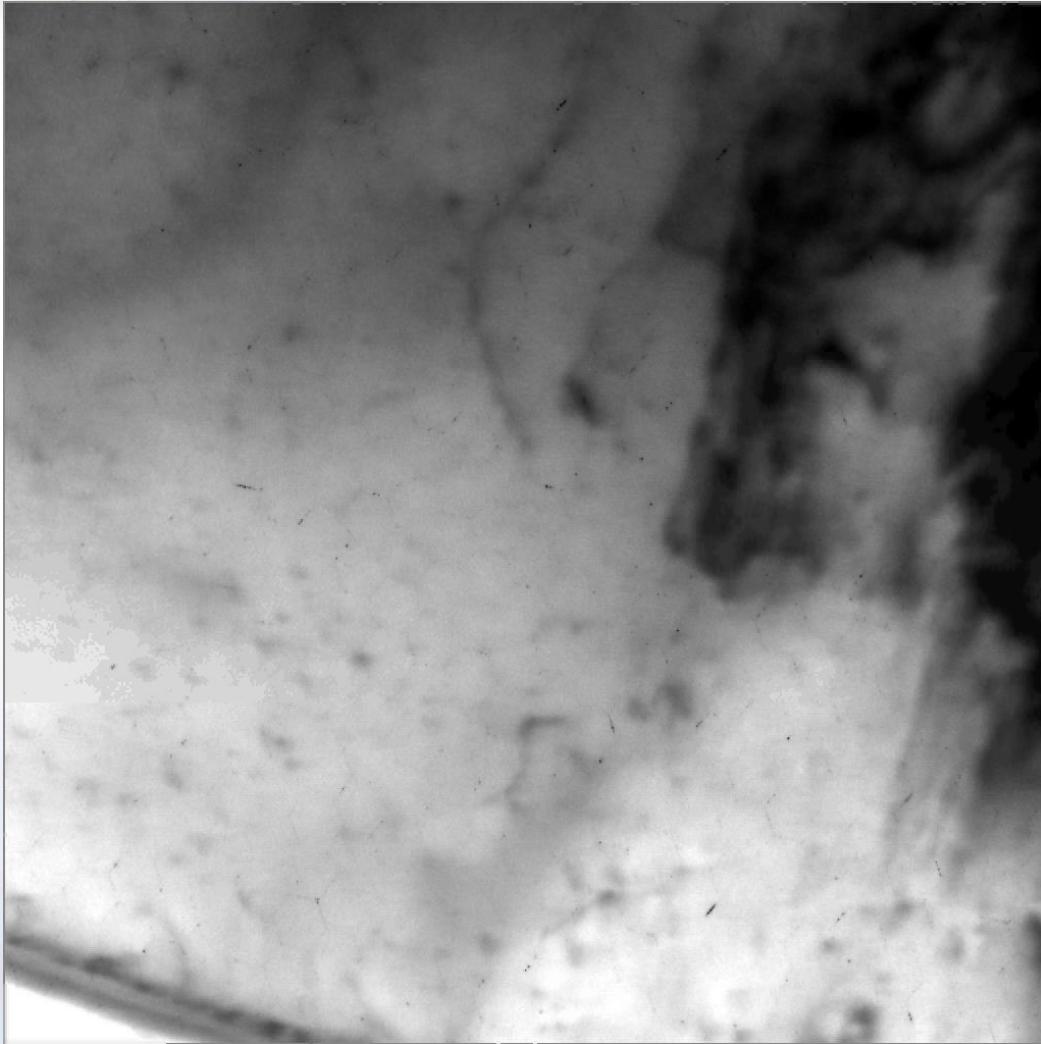
Ion species & energy introduced into the TEM



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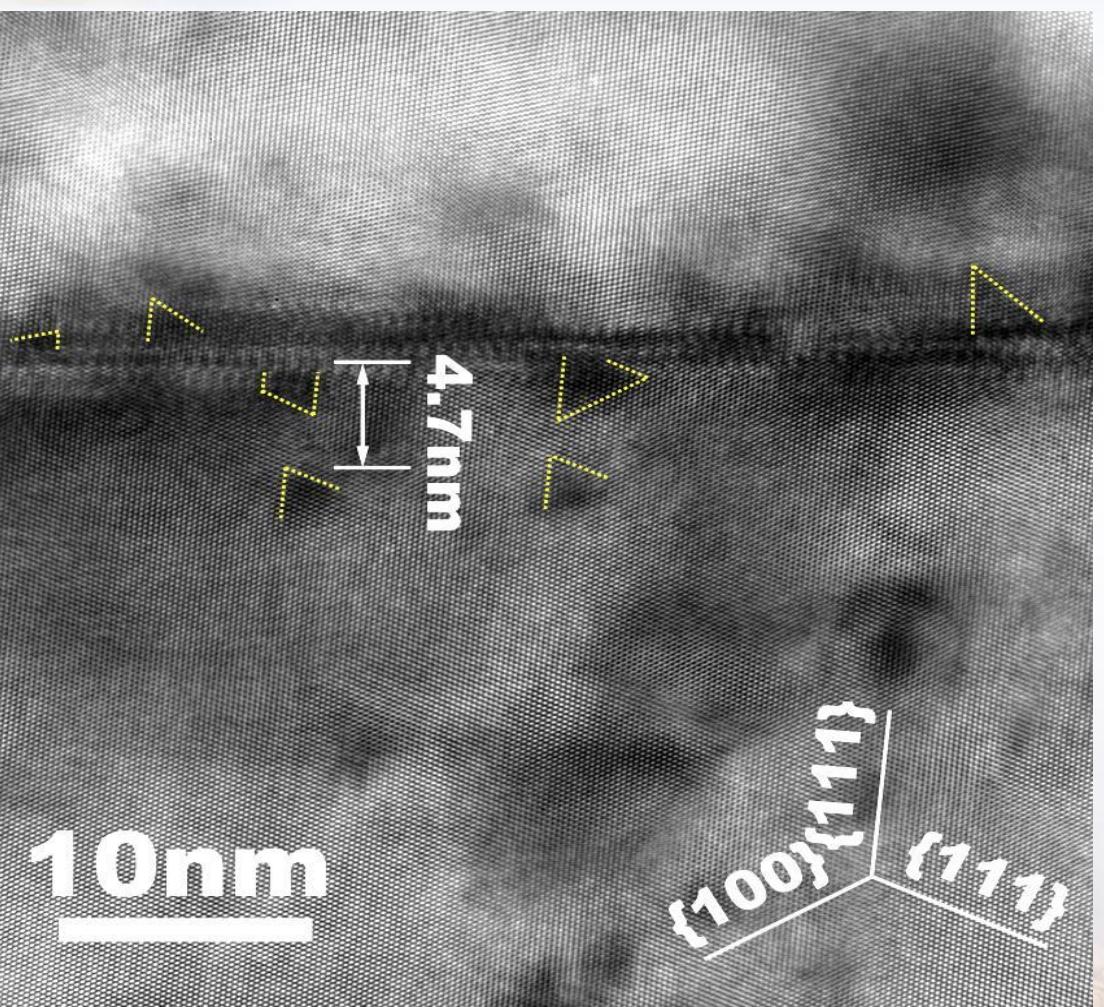
Quantifying Defect Evolution in Irradiated Cu

Collaborators: N. Li & A. Misra

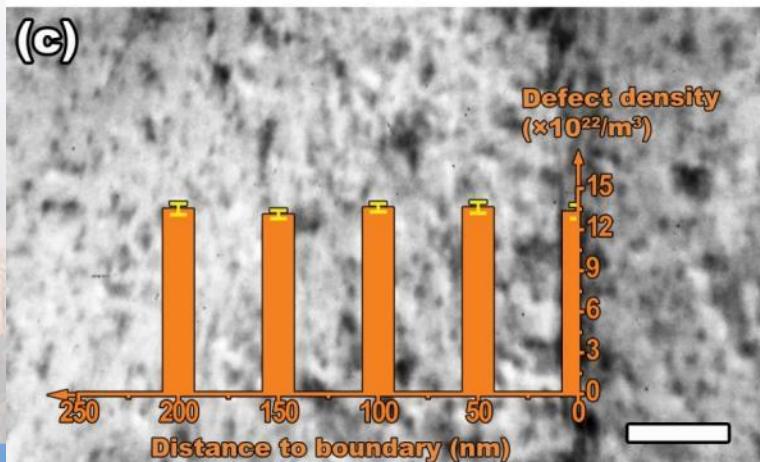
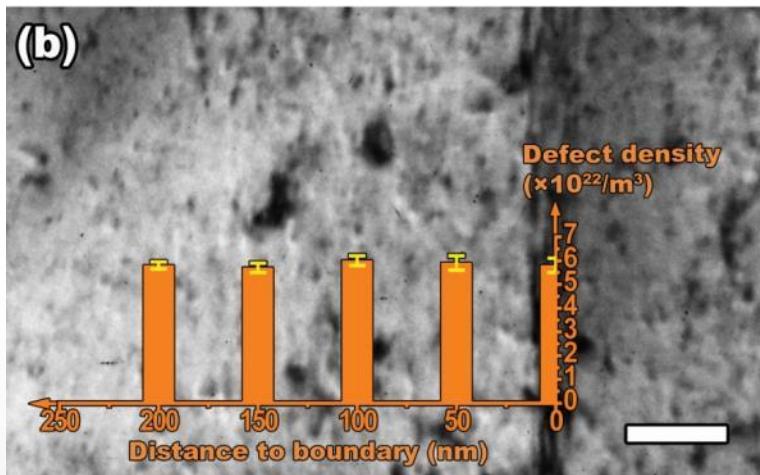
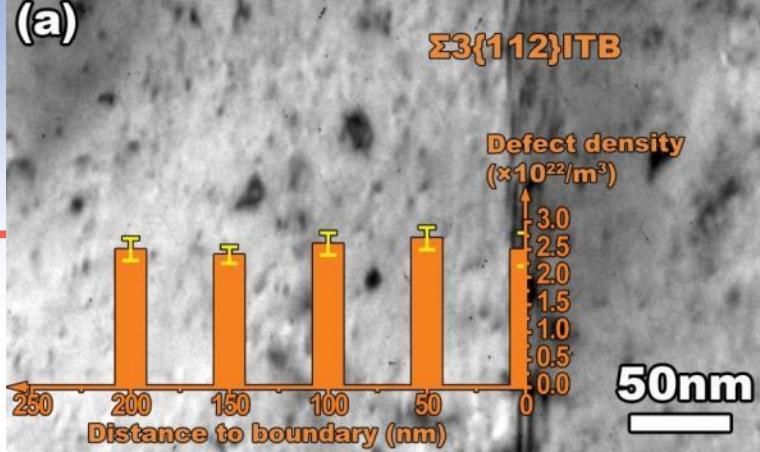


Defects are Altered Little by the Presence of Grain Boundaries

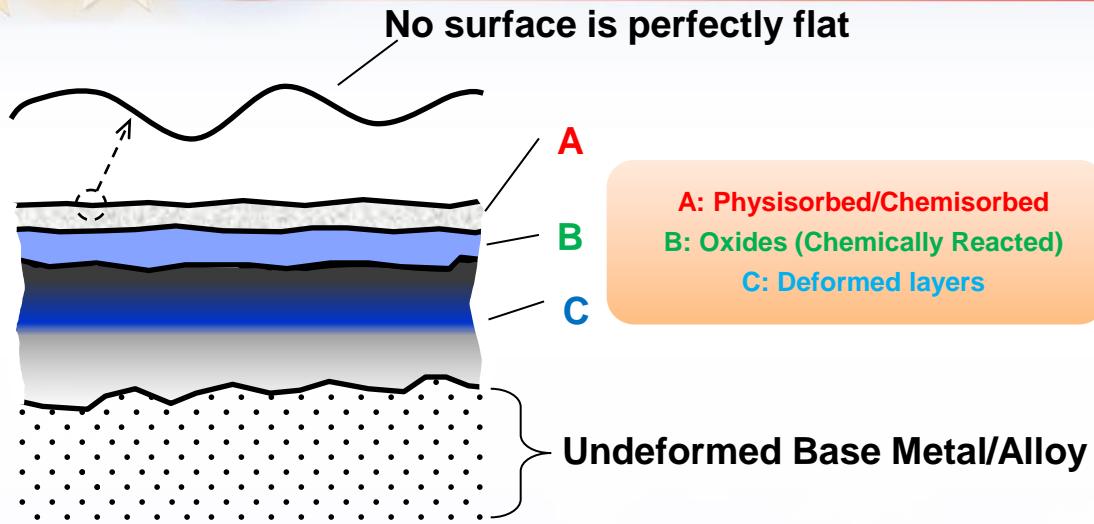
Collaborators: N. Li & A. Misra



SFT appear to be directly at GB
No change in defect density is observed near GB



Tailoring Wear Properties in Au Sliding Contacts

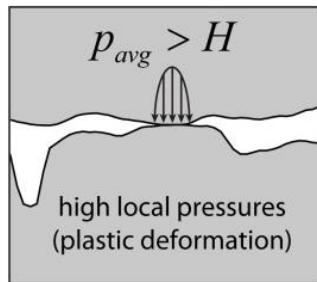


Real area of contact (A_r) to be minimized for low adhesion

(Low Adhesive Wear)

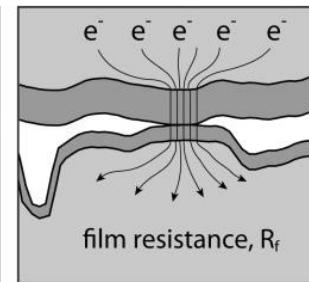
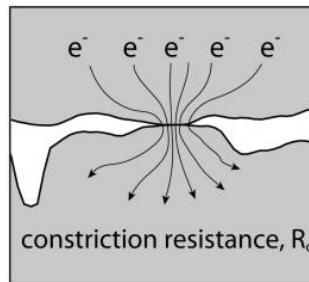
Or maximized for reduced electrical contact resistance (ECR)

Asperity Contacts, Constriction, Asperity Contacts and Surface Films
areal sum of asperity contacts and surface films define electrical contact resistance



... for metal contacts the real area is a function of hardness and contact force (Bowden & Tabor, 1939):

$$A_r \cong \frac{F_n}{H}$$



... ECR is a function of the constriction and film resistances:

$$ECR = \sum_i (R_{c,i} + R_{f,i})$$

Archard, *Journal of Applied Physics* (1953) 24:981

R. Holm, *Electrical Contacts Handbook* (1958) Berlin: Springer-Verlag

Greenwood & Williamson, *Proc. Royal Society* (1966) A295:300

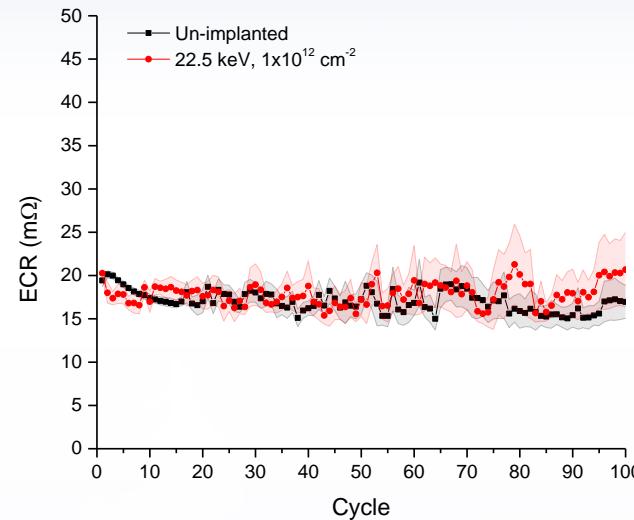
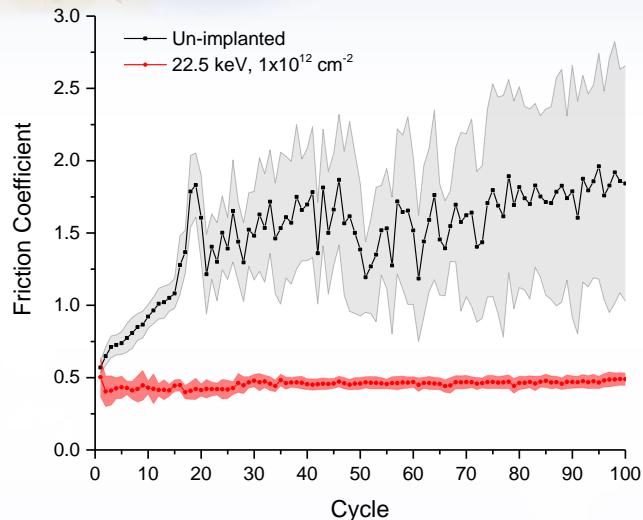
T.W. Scharf & S.V. Prasad, *Journal of Material Science* (2013) 48:511-531



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ECR and Wear Measurements

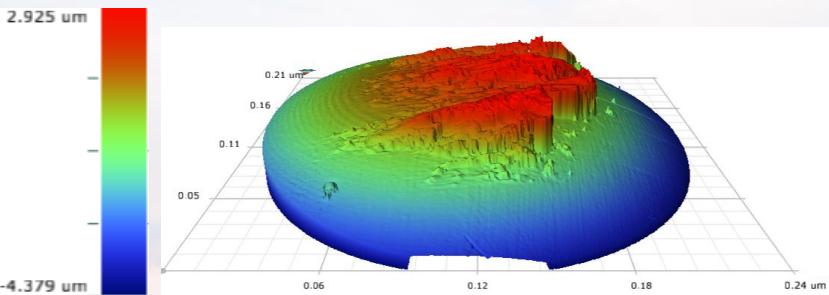
Collaborators: J-E Mogonye & S.V. Prasad



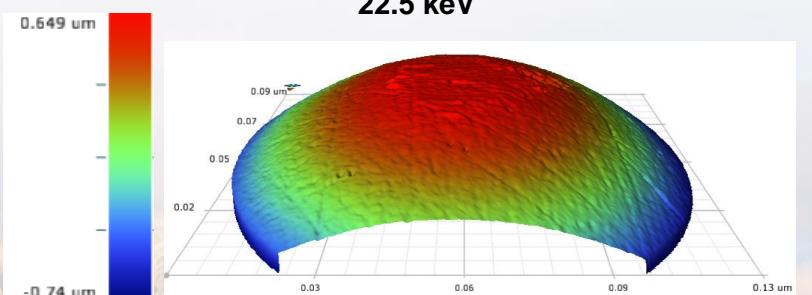
Friction is significantly reduced with ^3He implantation while maintaining ECR performance

Scanning white light interferometer topographical construction of riders after 100 cycles

Rider after 100 cycles against Un-implanted Au



Rider after 100 Cycles against Au implanted to $1\text{E}12 \text{ cm}^{-2}$ @ 22.5 keV



Wear is significantly reduced with minimal effect in ECR



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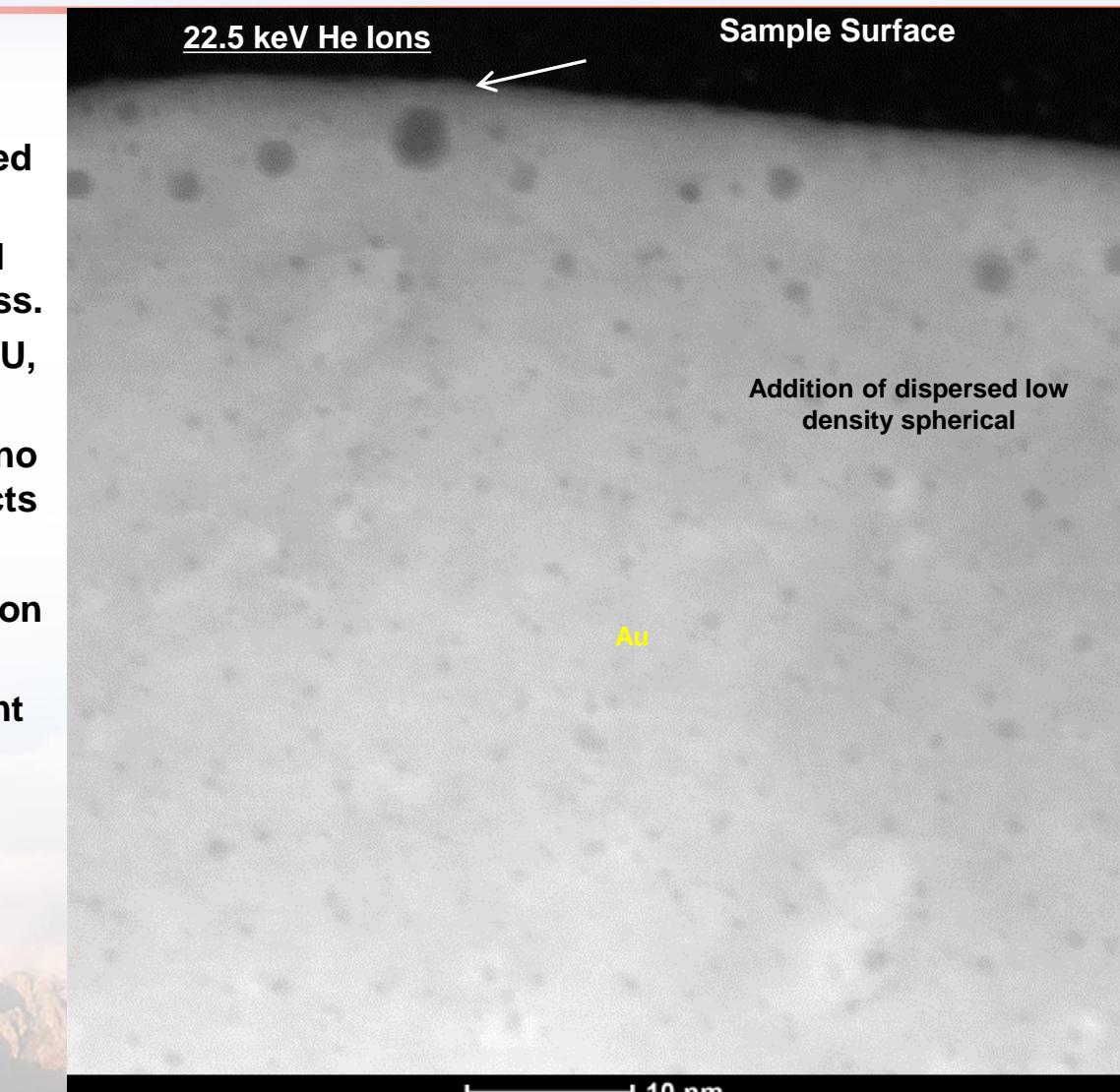
Modeling and STEM of He Implantation

Collaborators: J-E Mogonye & S.V. Prasad

- Simulations: SRIM 2008
- Monte-Carlo simulation of kinematic interaction based on empirical data fitted functions
 - Input variables of target material include density, AMU, and thickness.
 - Input variables of ions include AMU, energy, and angle of incidence.
 - Assumes isotropic material, thus no consideration for channeling effects
- AC-STEM used to observe the distribution of implanted bubbles
- Bubble locations are in good agreement with SRIM ion range predictions

He implantation result in small dispersed spherical structures assumed to be He bubbles.

Dispersion and depth can be tailored

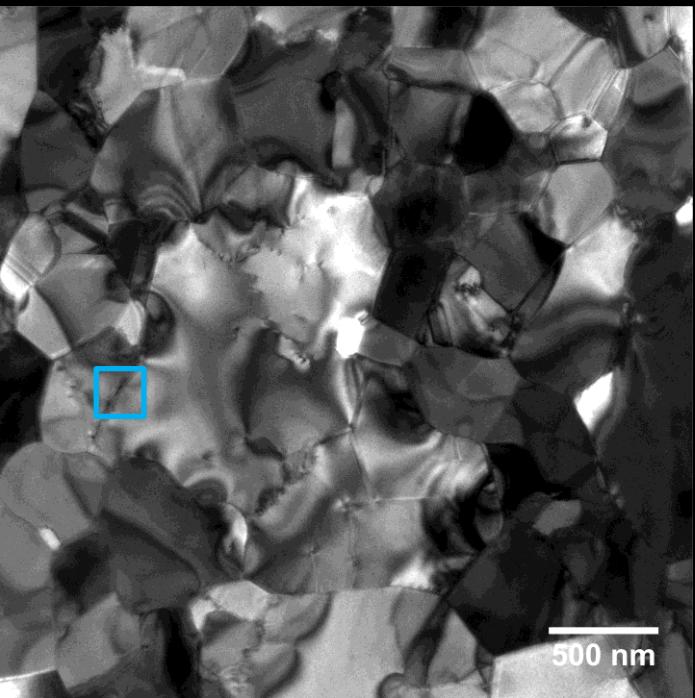


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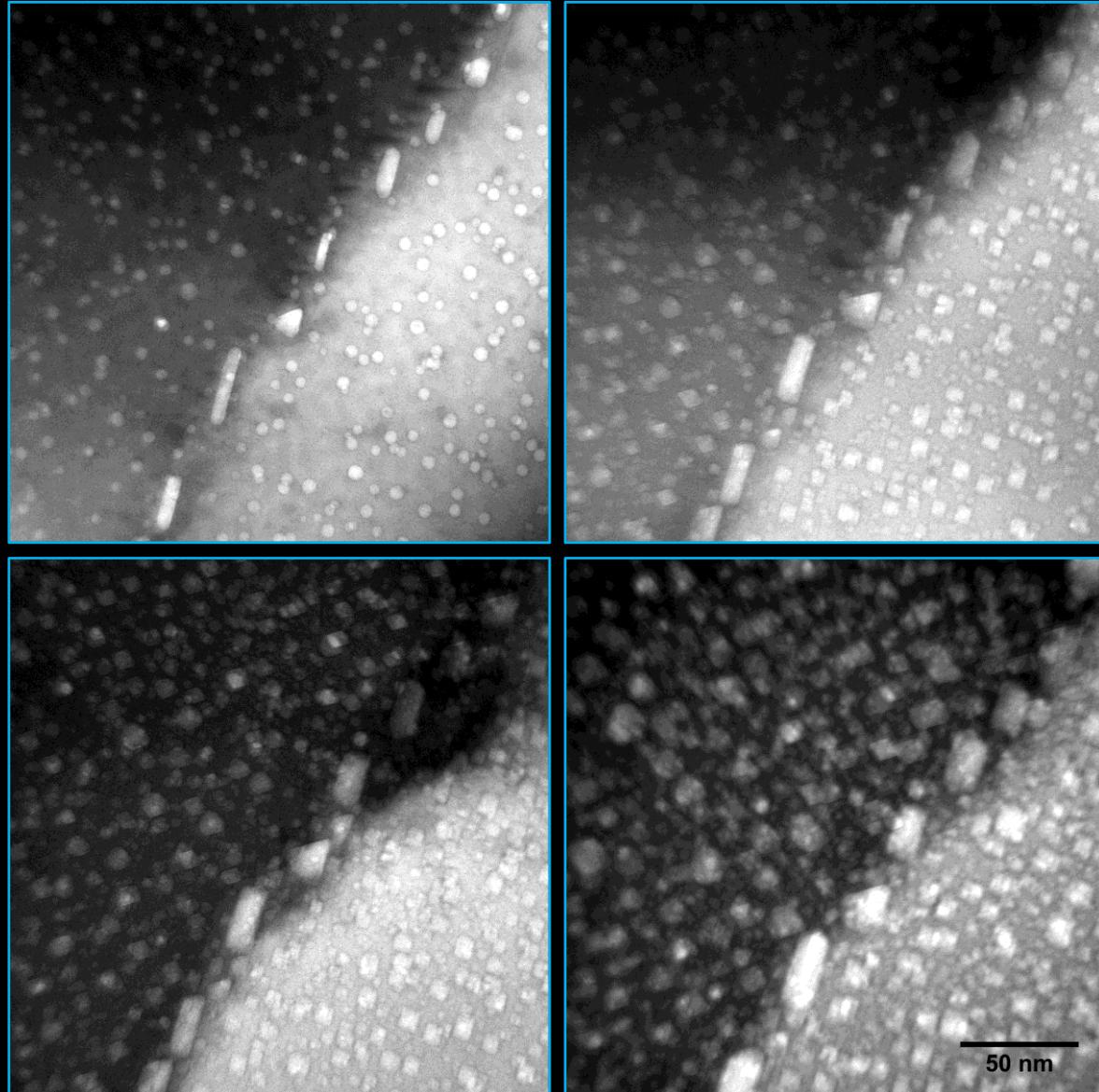
In situ Implantation

Collaborators: C. Chisholm & A. Minor

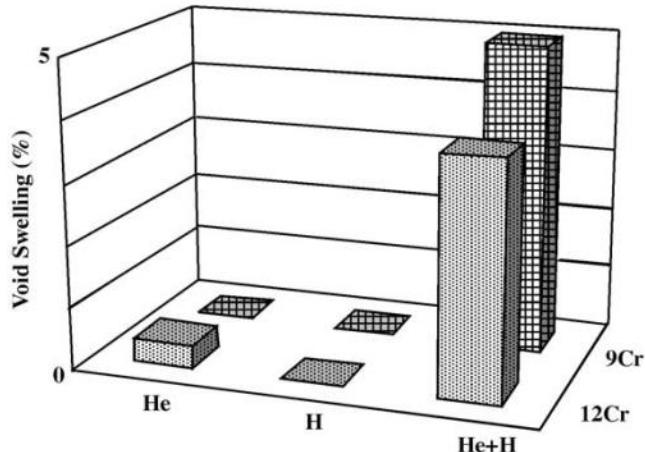
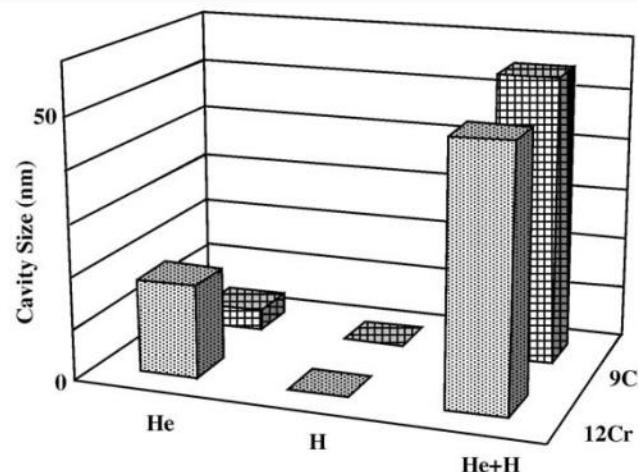
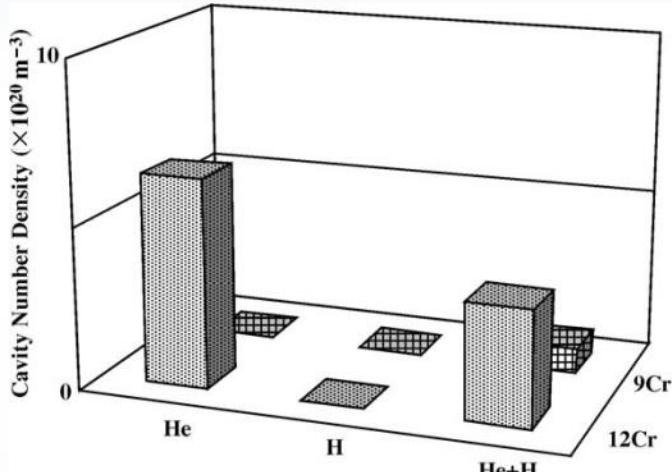


**Gold thin-film implanted
with 10keV He^{2+}**

**Result: porous
microstructure**



H, He, and Displacement Damage Synergy



T. Tanaka et al. "Synergistic effect of helium and hydrogen for defect evolution under multi-ion irradiation of Fe-Cr ferritic alloys"

J. of Nuclear Materials 329-333 (2004) 294-298

Coupling Effect

- H and He are produced as decay products
- The relationship between the point defects present, the interstitial hydrogen, and the He bubbles in the system that results in the increased void swelling has only been theorized.
- The mechanisms which governs the increased void swelling under the presence of He and H have never been experimental determined

No capability currently exist for triple beam irradiation in the U.S. and No capability for triple beam TEM ion irradiation exists in the world



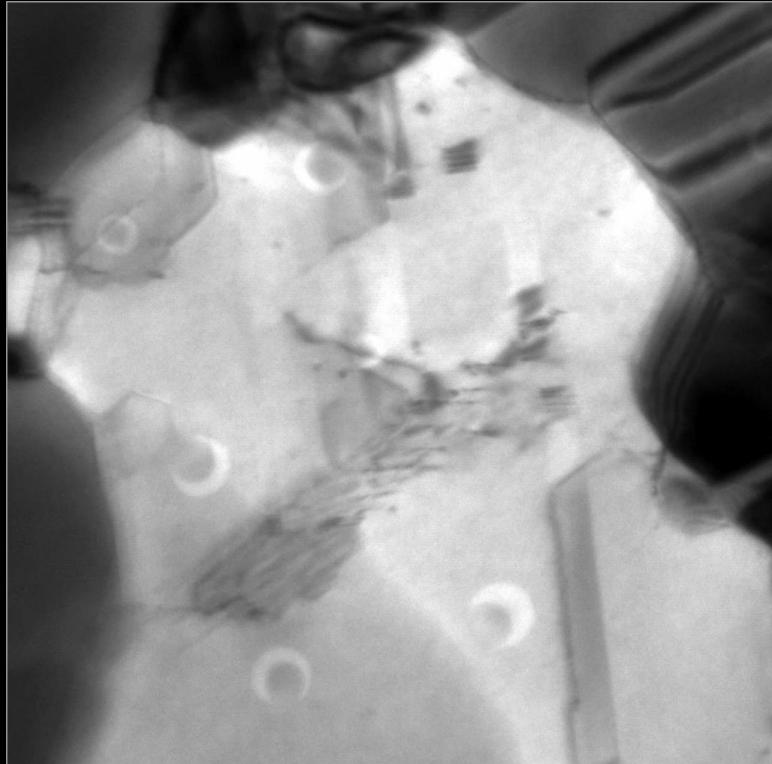
Sandia National Laboratories



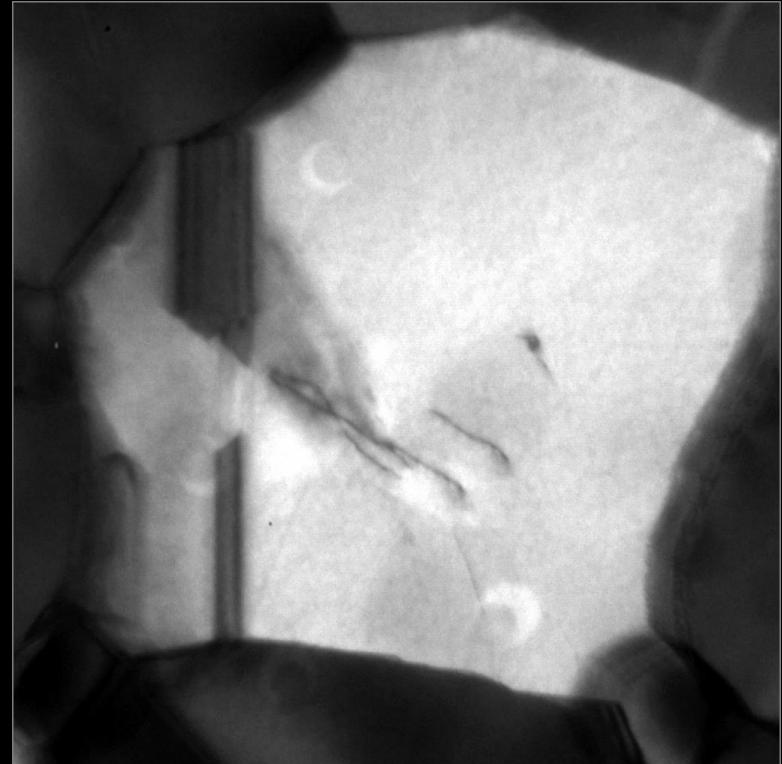
Single Ion Strikes

Collaborators: C. Chisholm & A. Minor

7.9×10^9 ions/cm²/s



6.7×10^7 ions/cm²/s



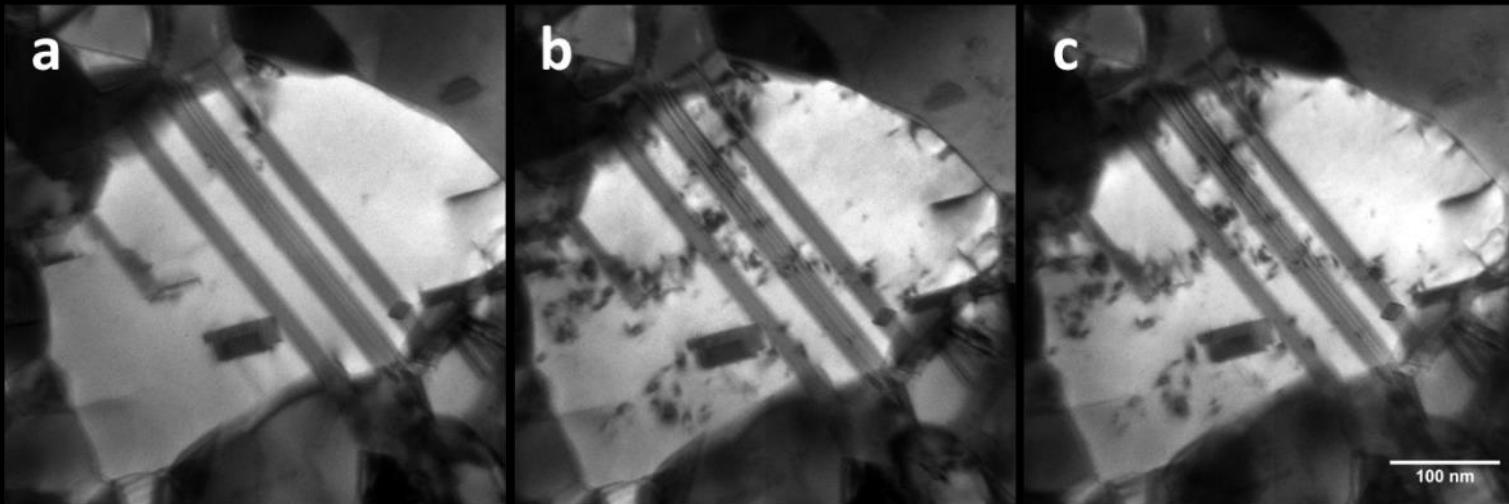
VS

Improved vibrational and ion beam stability permits us to work at 120kx or higher permitting imaging of single cascade events

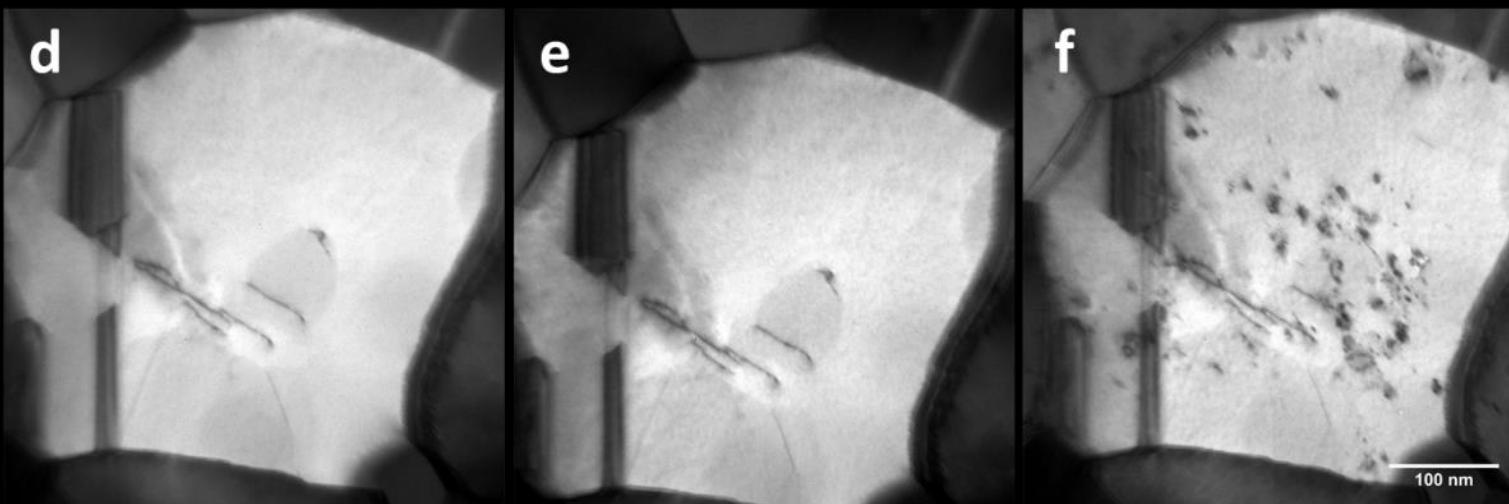
In situ Successive Implantation & Irradiation

Collaborators: C. Chisholm & A. Minor

Successive Au^{4+} then He^{1+}

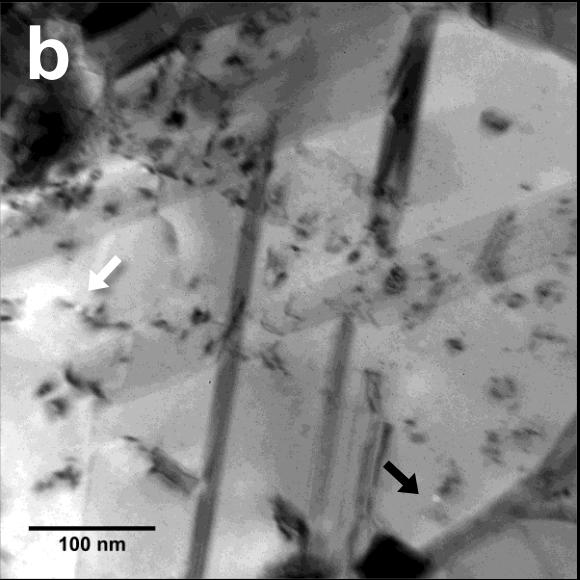
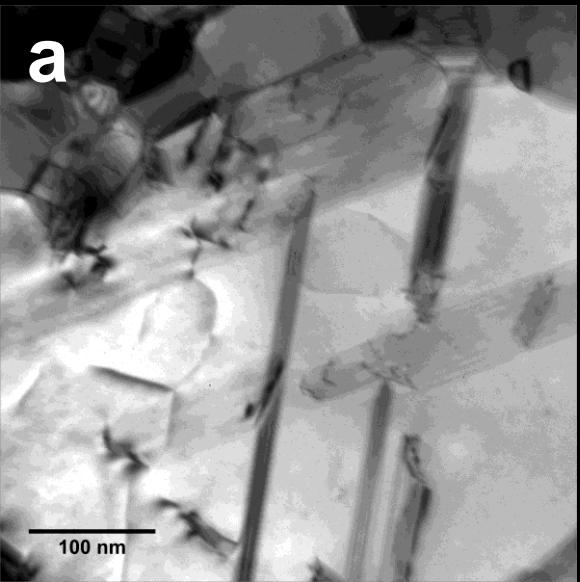


Successive He^{1+} then Au^{4+}

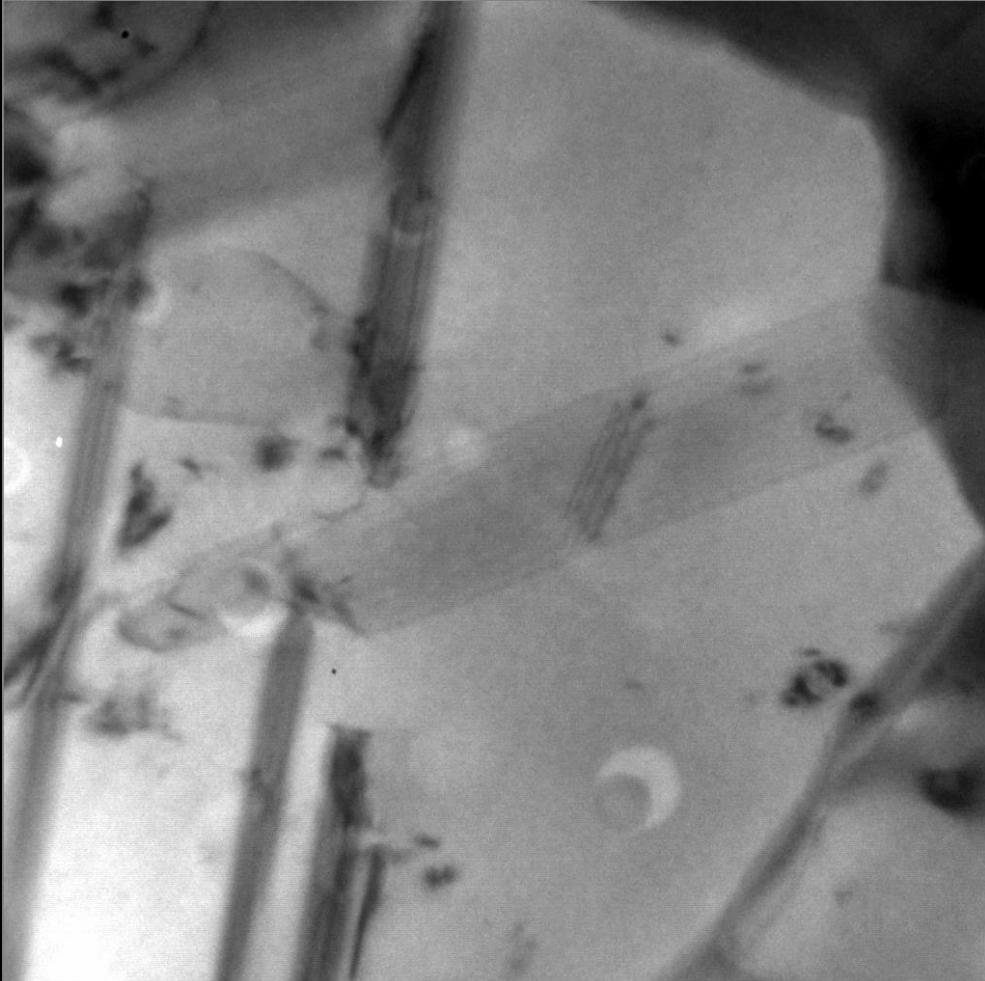


In situ Concurrent Implantation & Irradiation

Collaborators: C. Chisholm & A. Minor



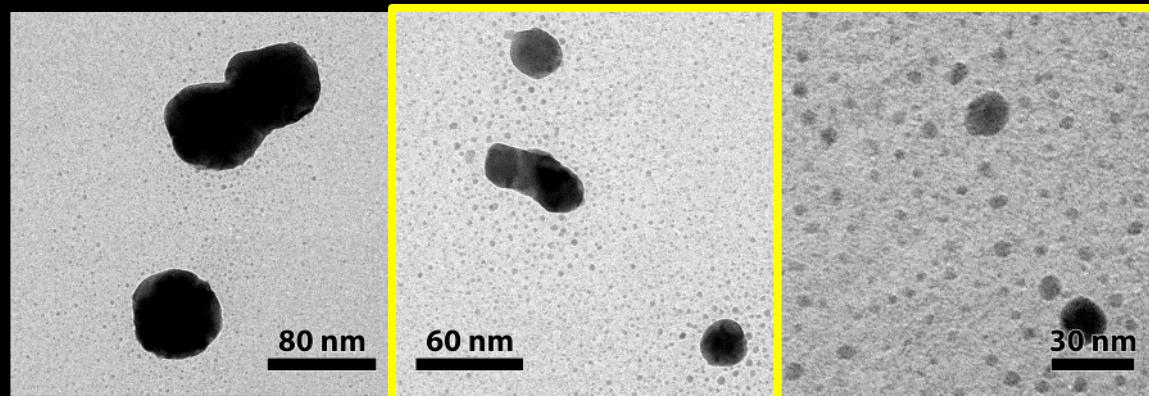
He^{1+} implantation and Au^{4+} irradiation
of a gold thin film



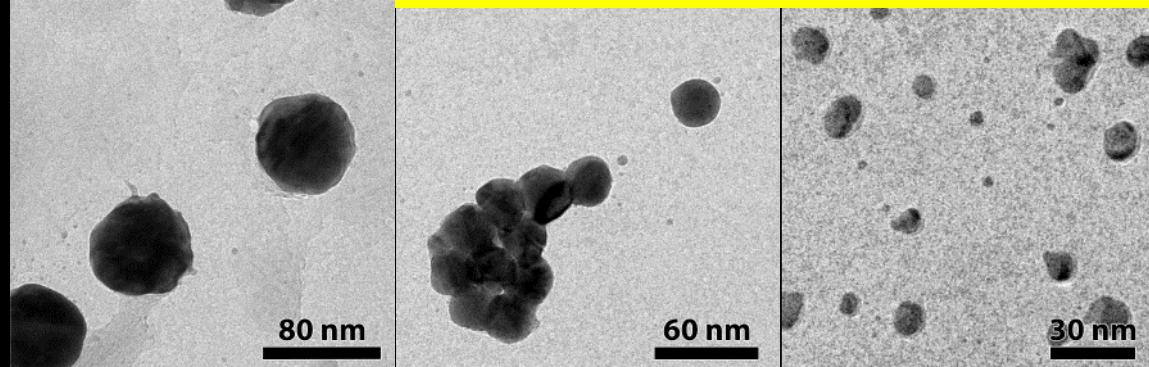
Cumulative Effects of Ion Irradiation as a Function of Ion Energy and Au Particle Size

Collaborator: D.C. Bufford

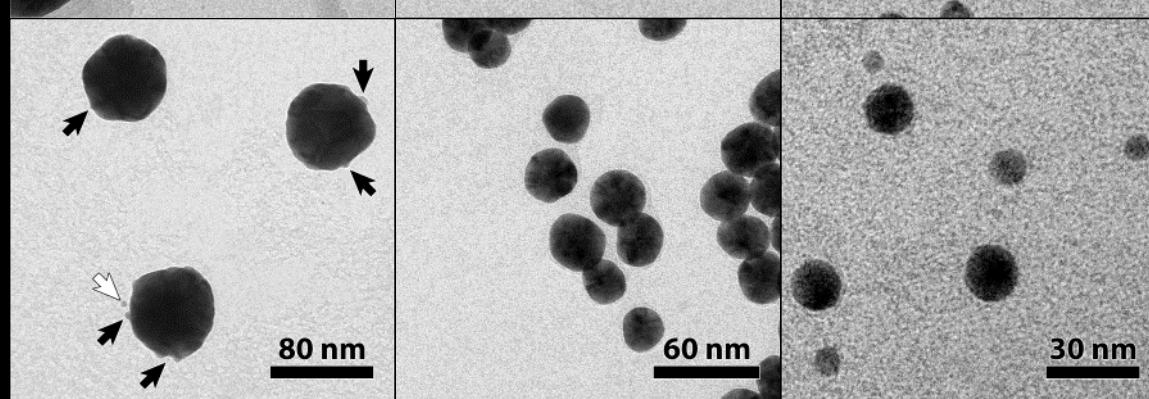
46 keV Au¹⁻
 $3.4 \times 10^{14} /cm^2$



2.8 MeV Au⁴⁺
 $4 \times 10^{13} /cm^2$



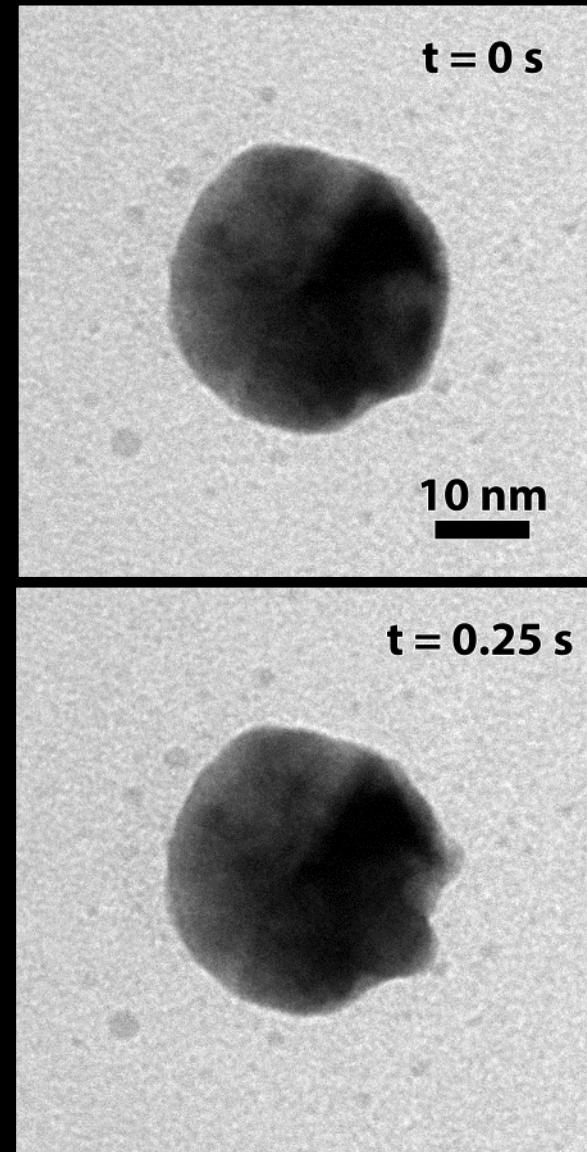
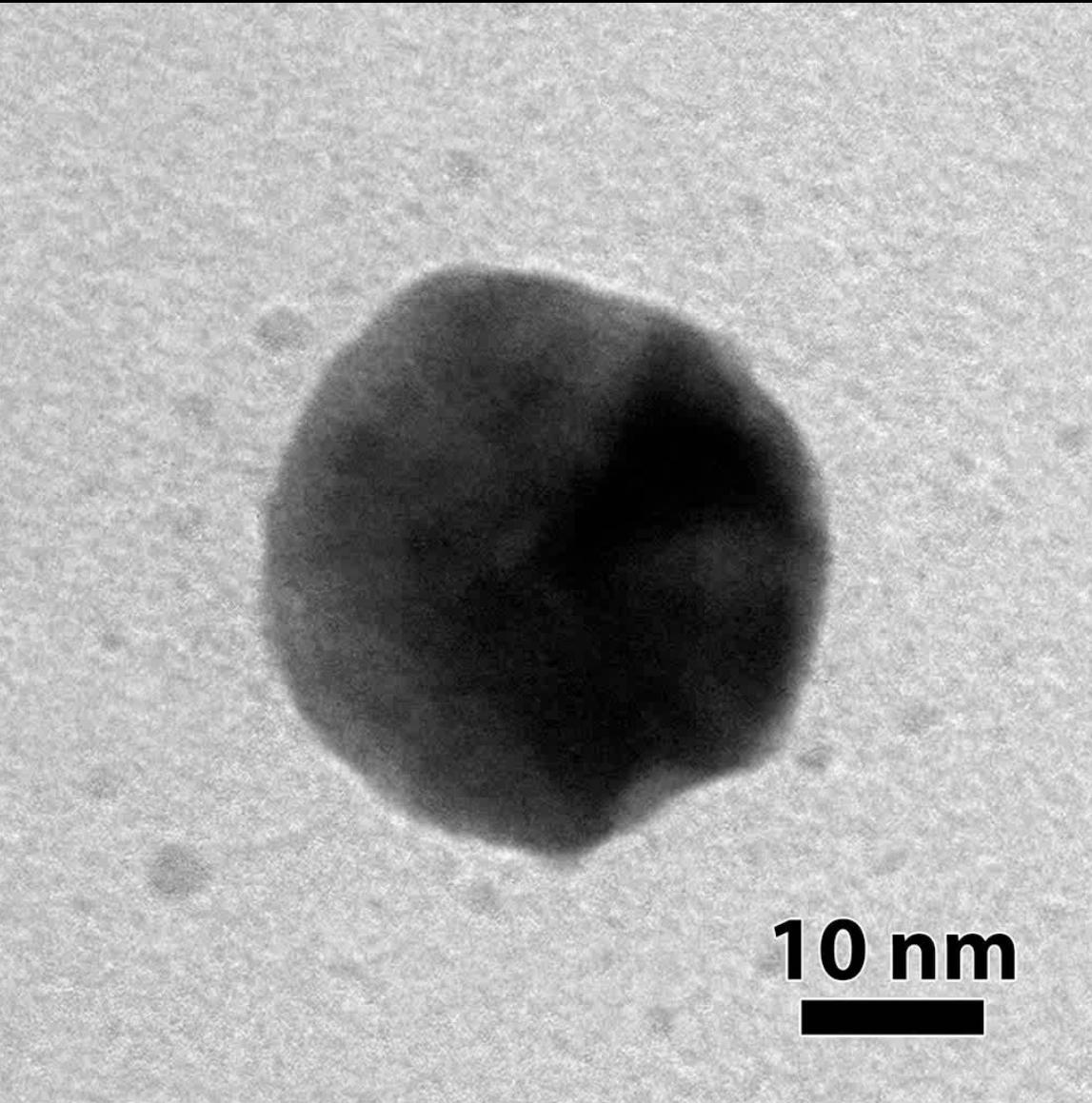
10 MeV Au⁸⁺
 $1.3 \times 10^{12} /cm^2$



Particle and ion energy dictate the ratio of sputtering, particle motion, particle agglomeration, and other active mechanisms

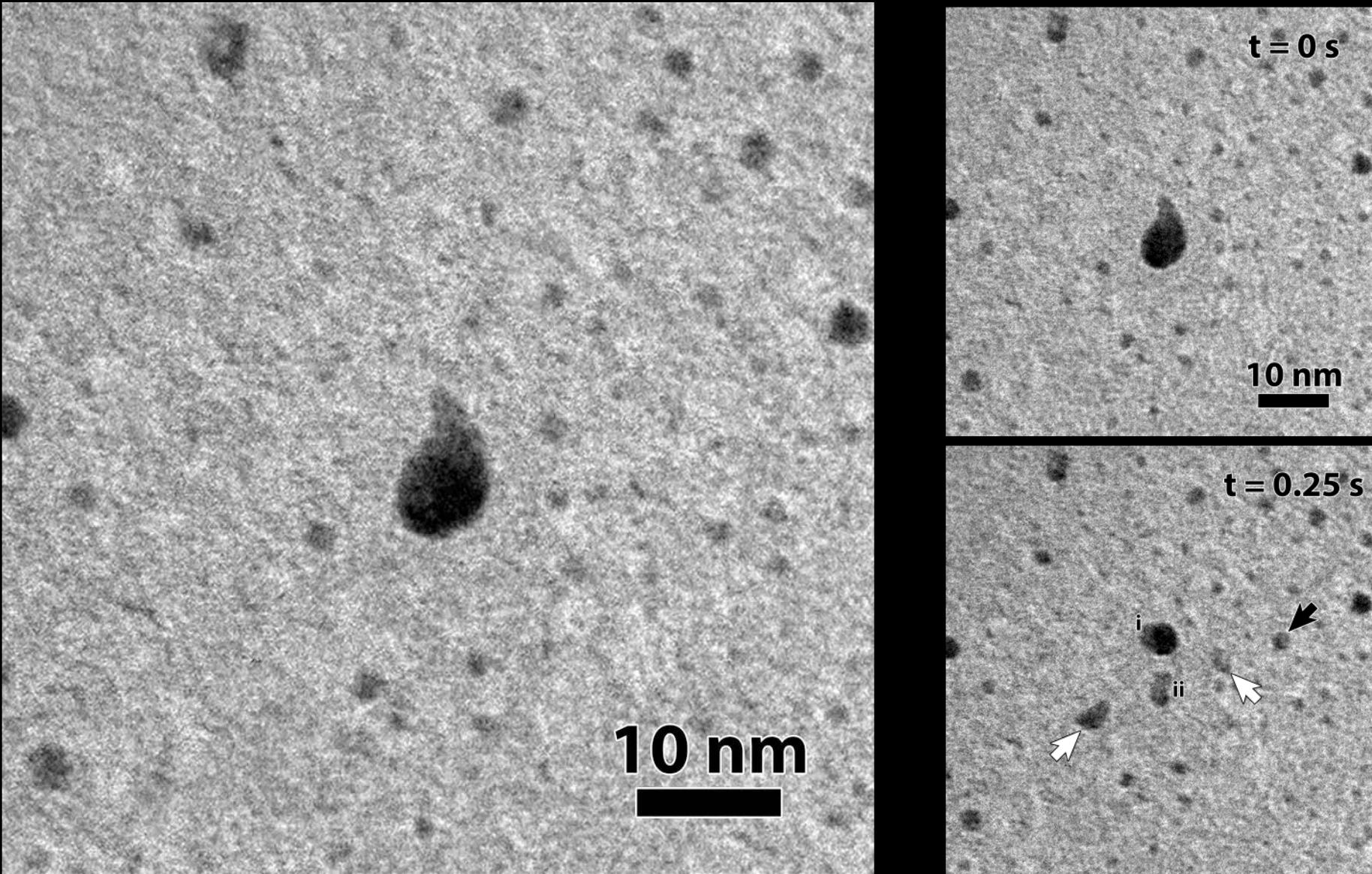
Single Ion Effects with 46 keV Au¹⁻ ions: 20 nm

Collaborator: D.C. Bufford



Single Ion Effects with 46 keV Au¹⁻ ions: 5 nm

Collaborator: D.C. Bufford



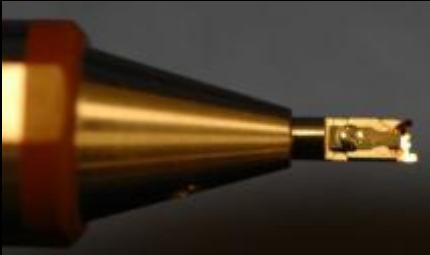
Advanced Microscopy Techniques Applied to Nanoparticles in Radiation Environments

Collaborators: S.M. Hoppe & T.J. Boyle

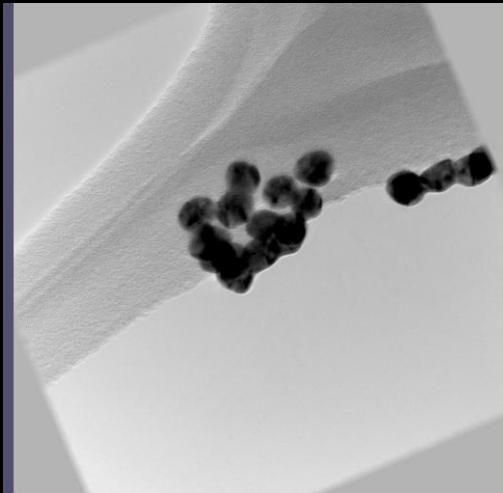
In situ Ion Irradiation TEM (I³TEM)



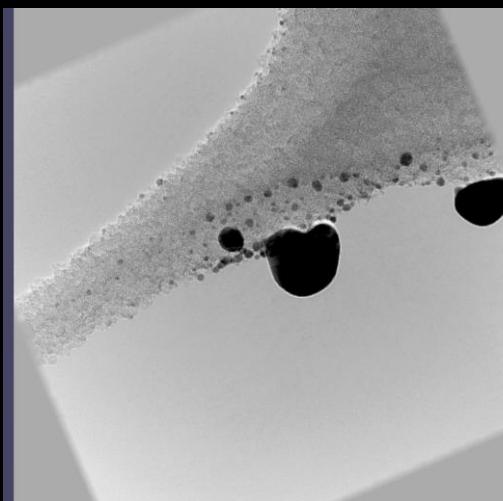
Hummingbird
tomography stage



Aligned Au NP tilt series -
unirradiated



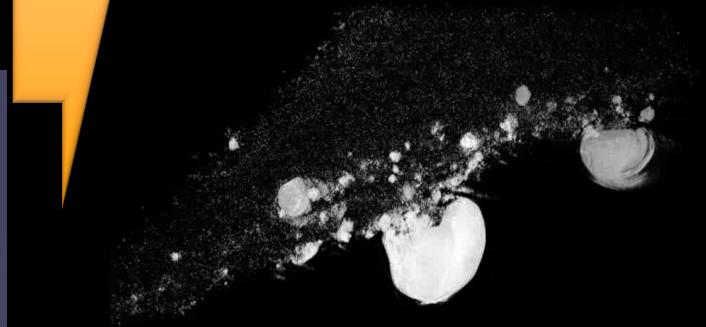
Aligned Au NP tilt series -
irradiated



Unirradiated Au NP model



Irradiated Au NP model

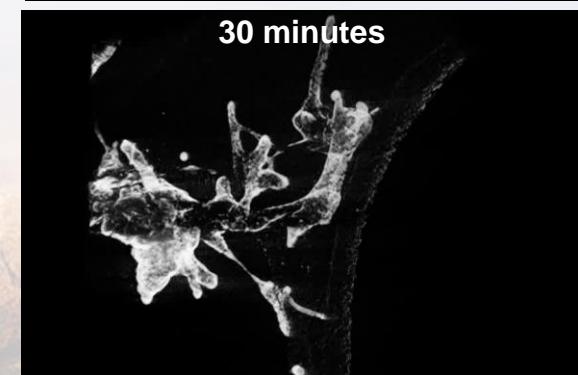
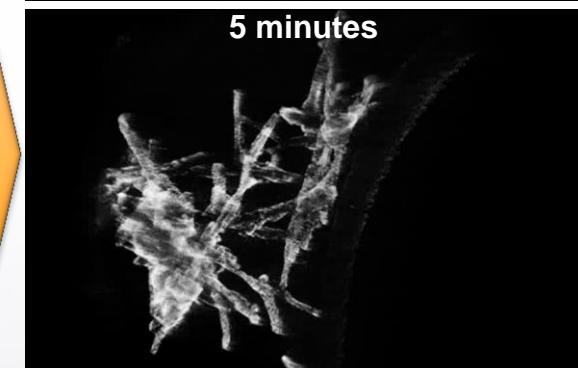
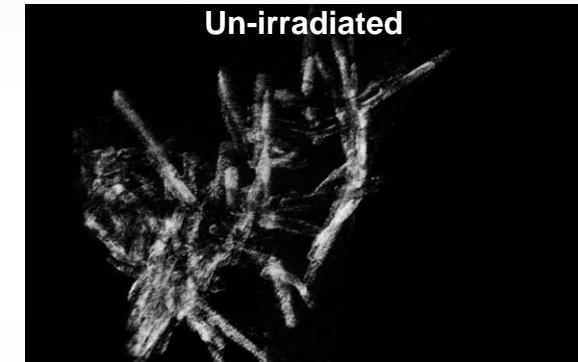


The application of advanced
microscopy techniques to
extreme environments provides
exciting new research directions

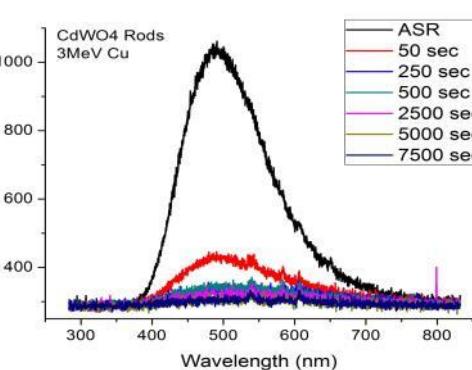
Radiation Tolerance is Needed in Advanced Scintillators for Non-proliferation Applications



In situ Ion Irradiation TEM (I³TEM)



High-Z nanoparticles (CdWO_4) are promising, but are radiation sensitive



+

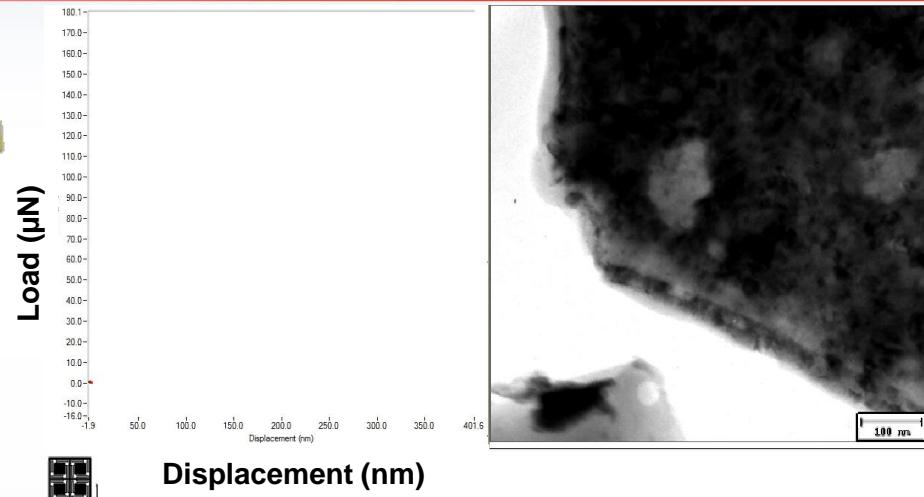
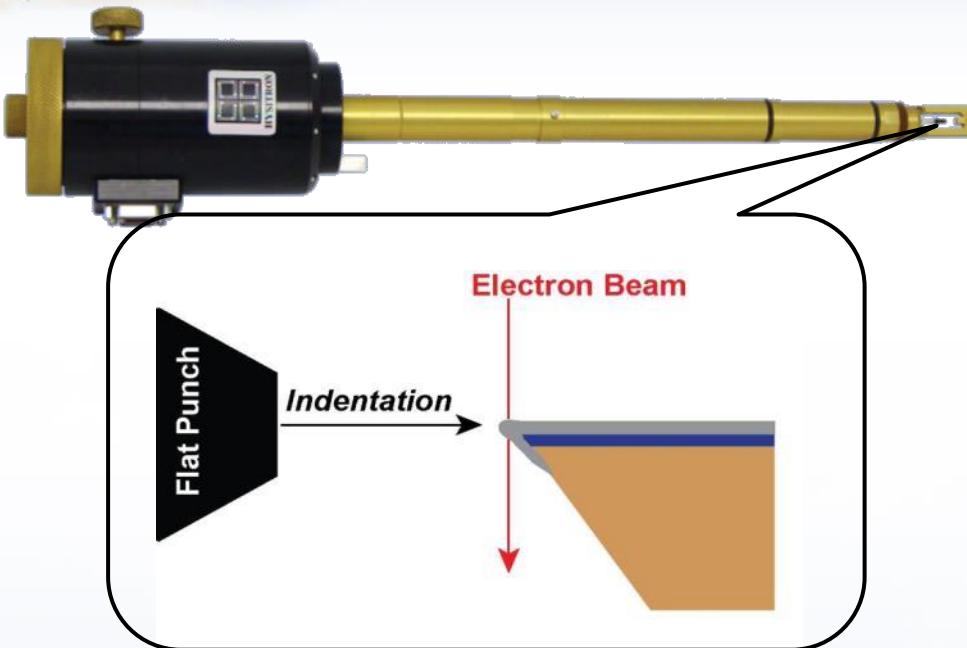
Hummingbird tomography stage



Tomography of Irradiated CdWO₄:
3 MeV Cu³⁺ at ~30 nA

In situ TEM Quantitative Mechanical Testing

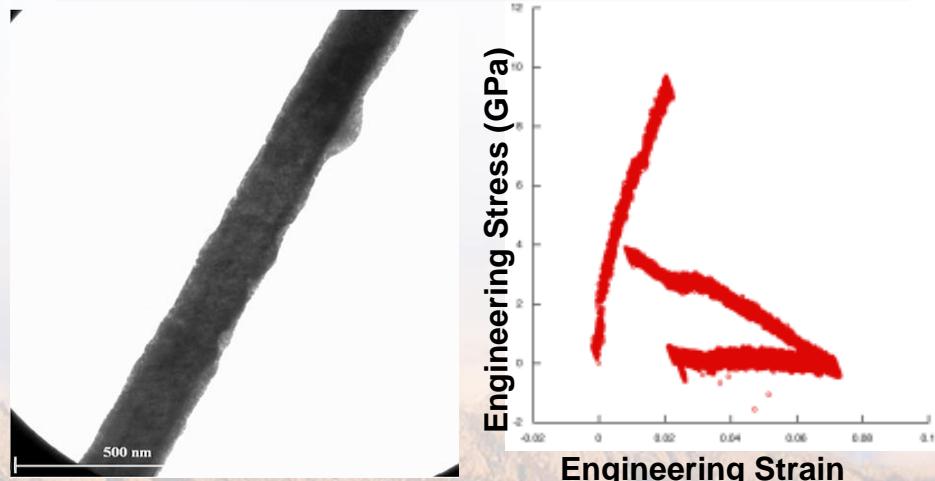
Contributors: J. Sharon, B. L. Boyce, C. Chisholm, H. Bei, E.P. George, P. Hosemann, A.M. Minor, & Hysitron Inc.



Fundamentals of Mechanical Properties

Range of Mechanical Testing Techniques

- Indentation
- Compression
- Tension
- Bending
- Wear
- Fatigue
- Creep



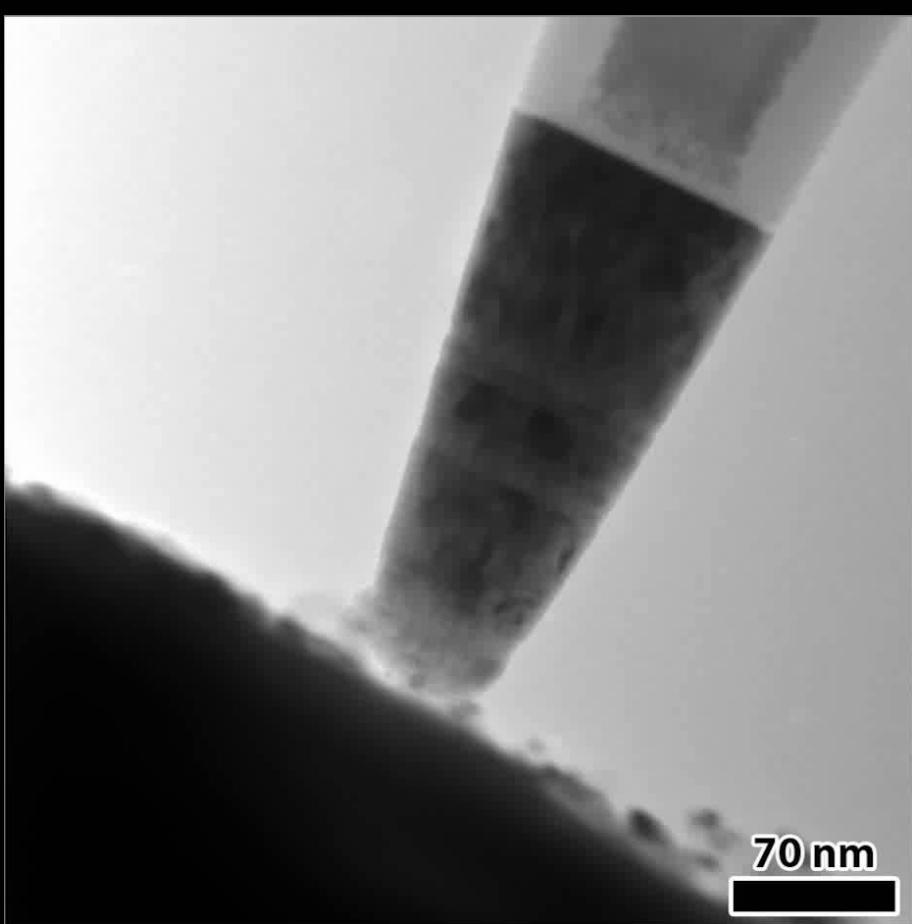
We have started looking at the effects of ion irradiation on mechanical properties



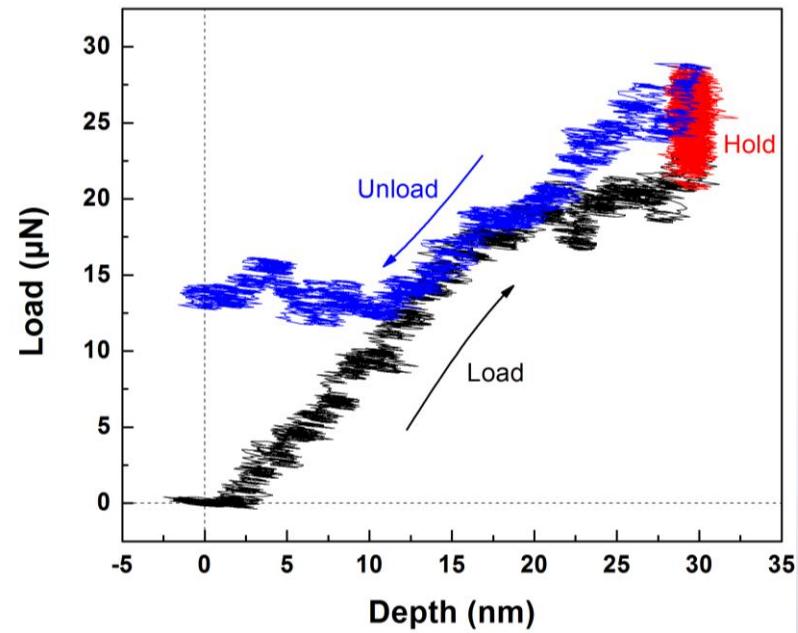


NC Ni Pillar Indentation

Collaborator: D.C. Bufford & W.M. Mook



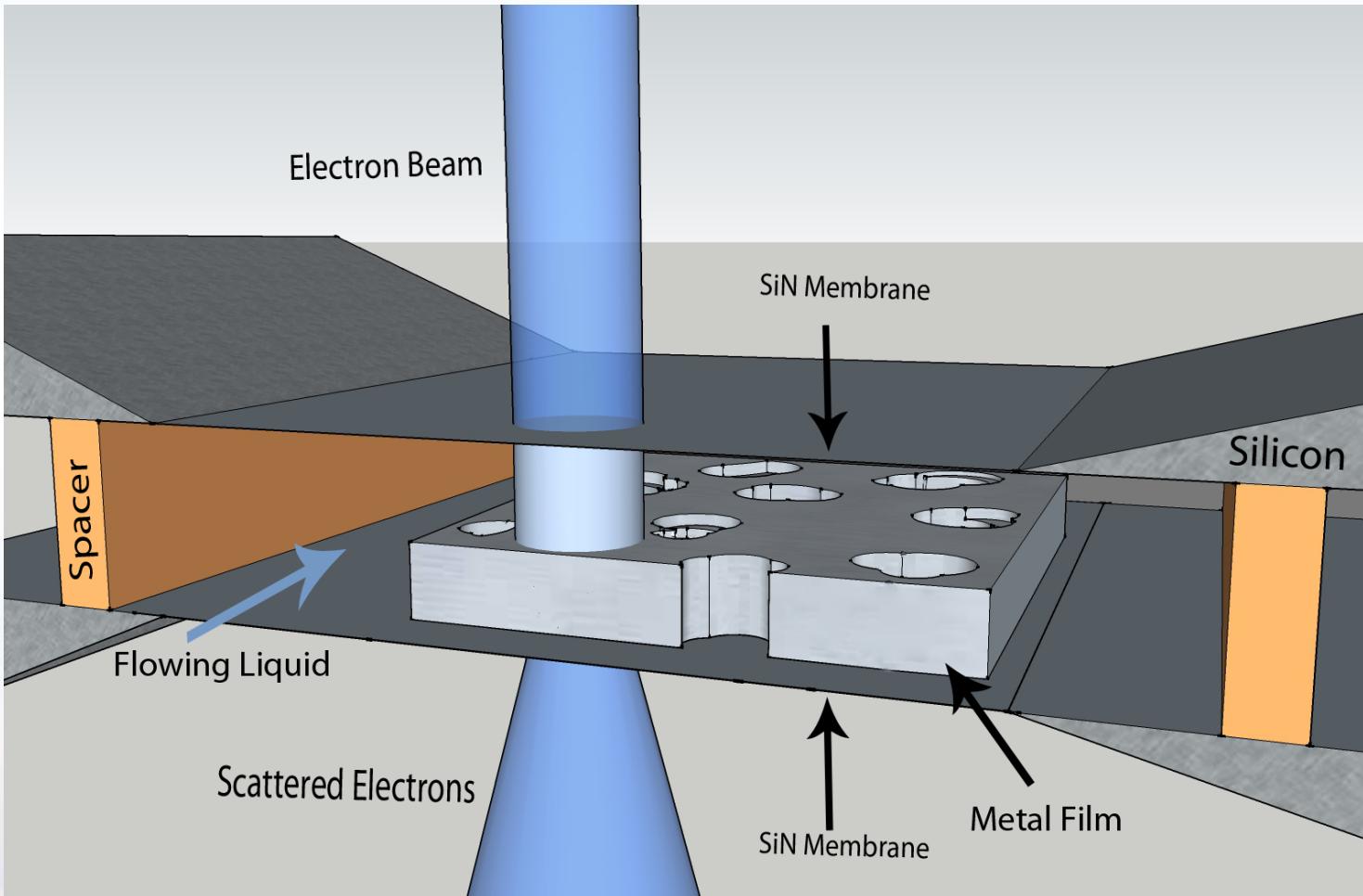
- 0.5 nm/s loading rate
- Trapezoid load function
- 60s load/60s hold/60s unload



Can We Gain Insight into the Corrosion Process through *In situ* TEM?

Microfluidic Stage

- Mixing of two or more channels
- Continuous observation of the reaction channel
- Chamber dimensions are controllable
- Films can be directly deposited on the electron transparent SiN membrane

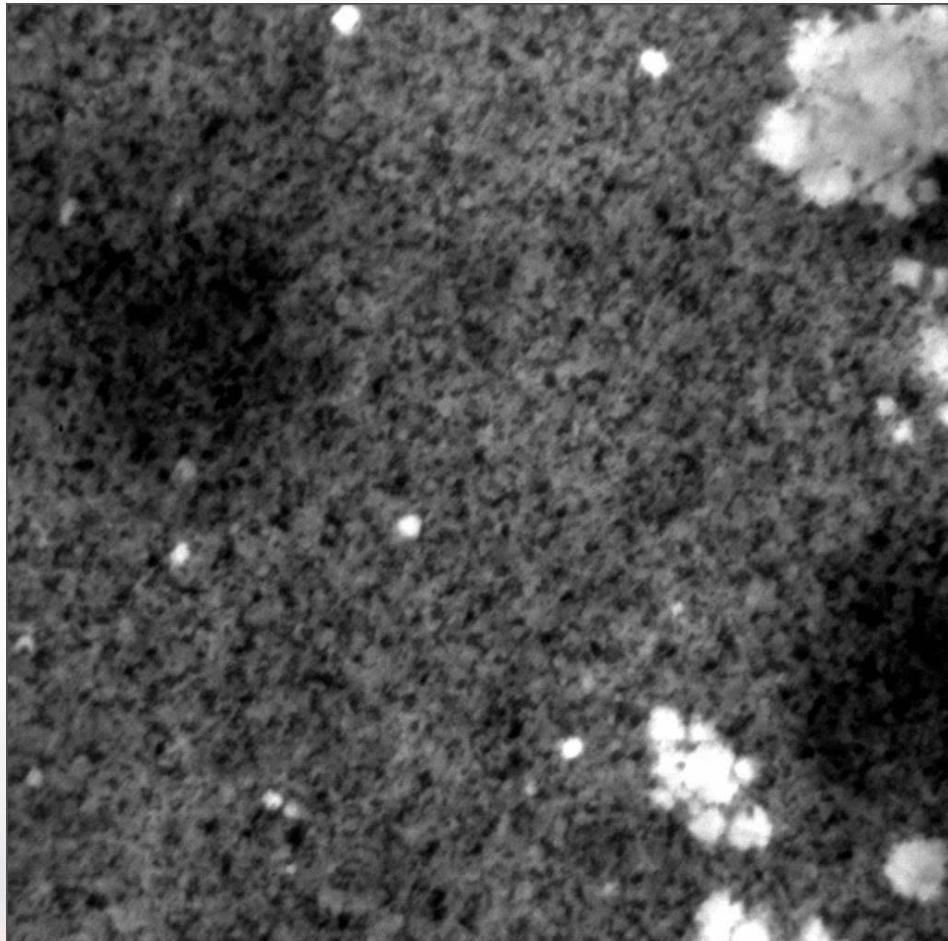
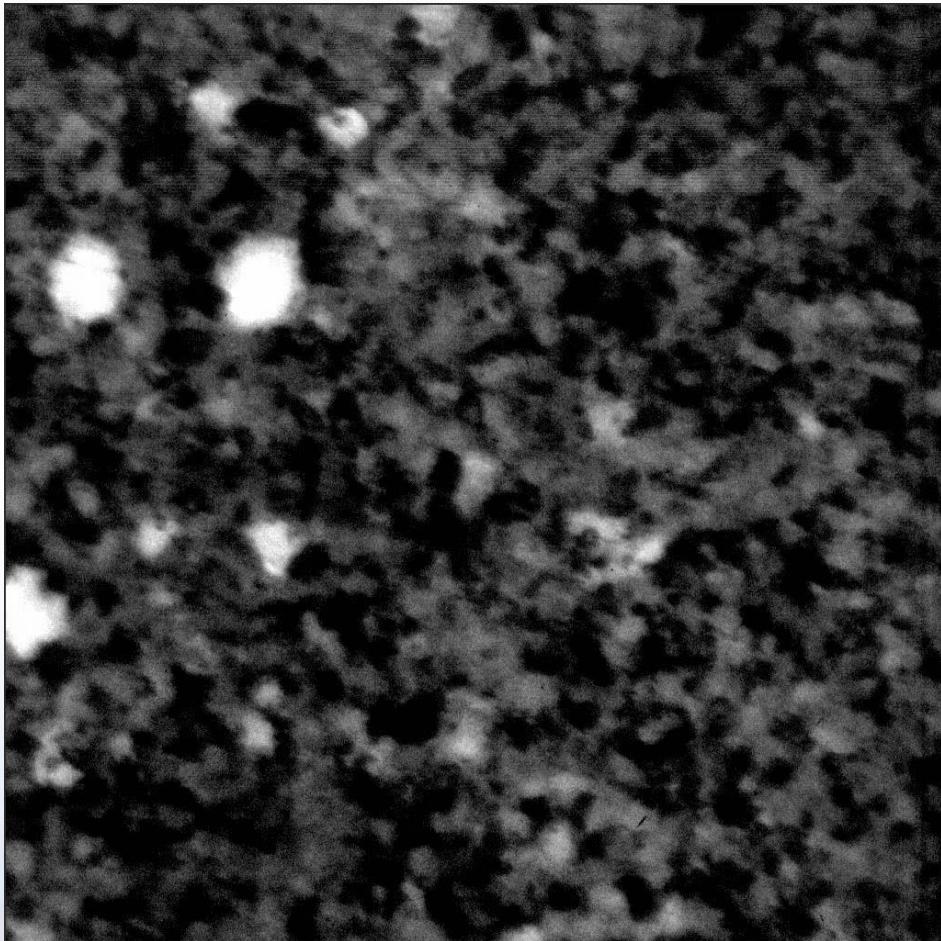


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Acetic Acid Corroding Nanograined Iron

Collaborators: D. Gross, J. Kacher, & I.M. Robertson



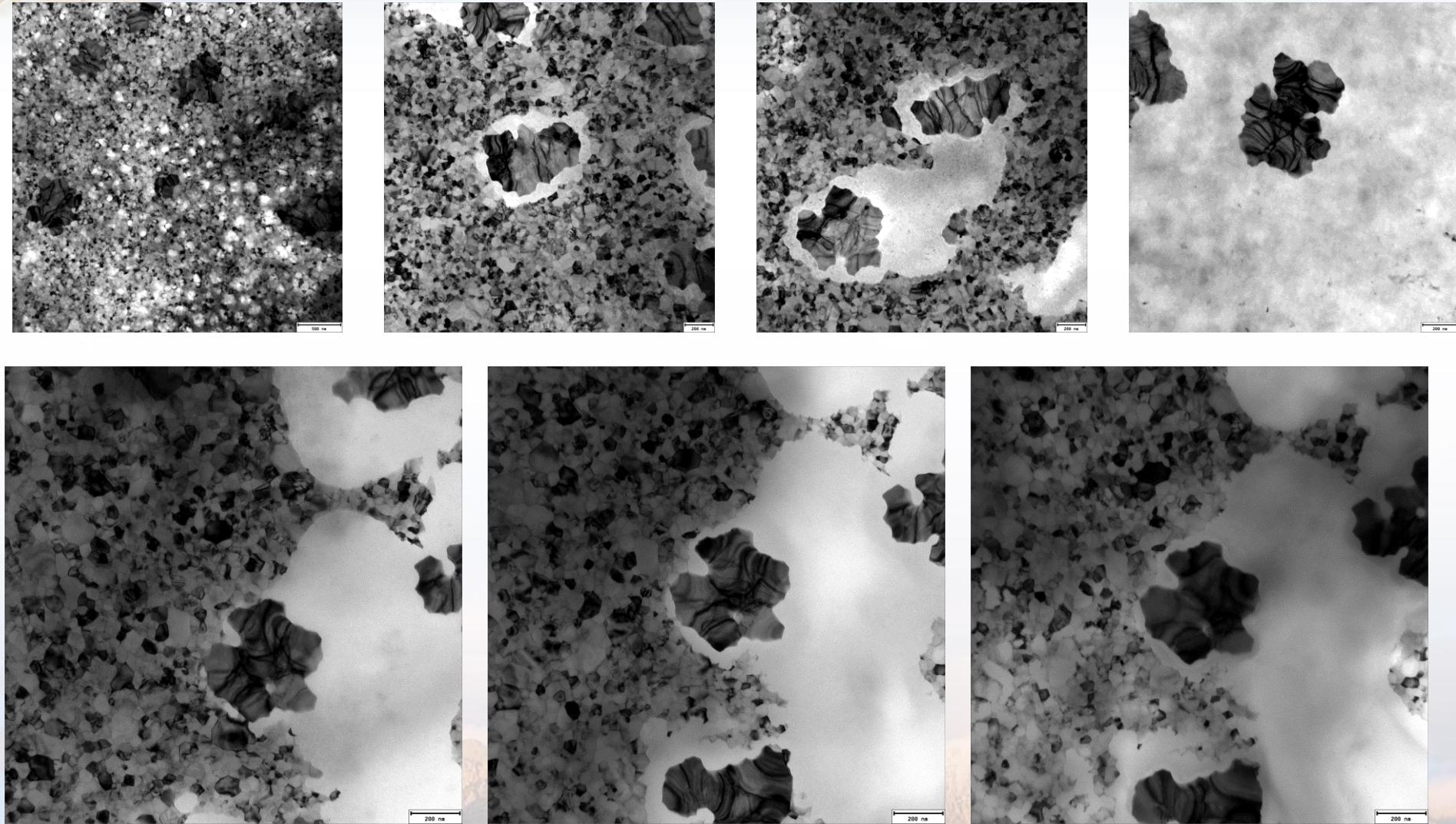
Pitting mechanisms during dilute flow of acetic acid over 99.95% nc-PLD Fe involves many grains.



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Acetic Acid Corroding in Annealed Nanograined Iron

Collaborators: D. Gross, J. Kacher, & I.M. Robertson



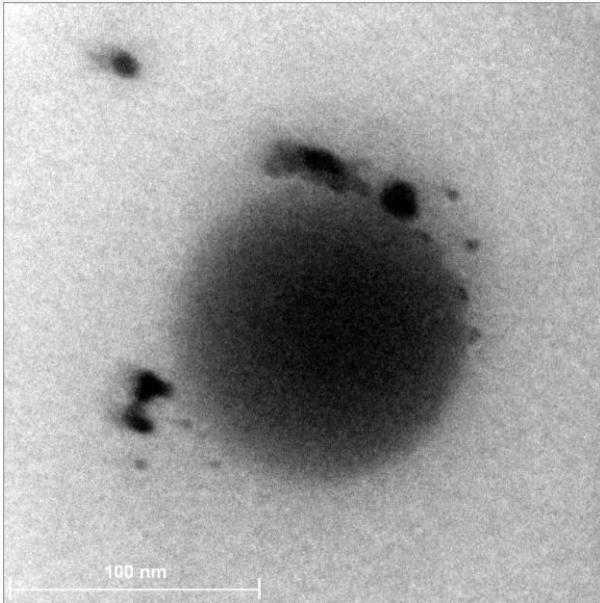
Large grains resulting from annealing appear more corrosion tolerant

Other Fun Uses of Microfluidic Cell

Protocell Drug Delivery

S. Hoppe,
E. Carnes,
J. Brinker

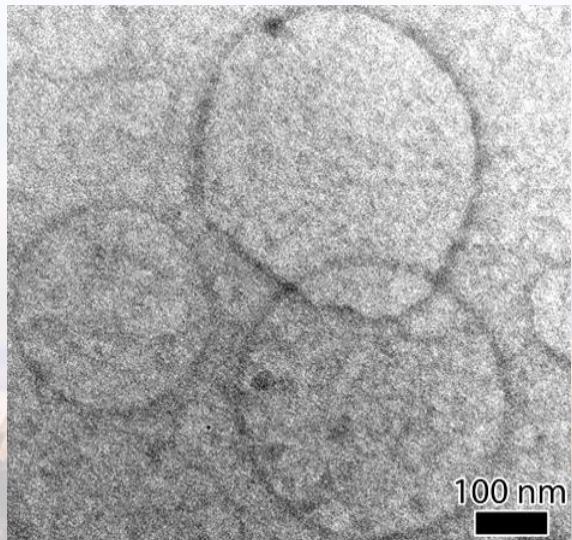
Liposome
encapsulated
Silica destroyed
by the electron
beam



Liposomes in Water

S. Hoppe,
D. Sasaki

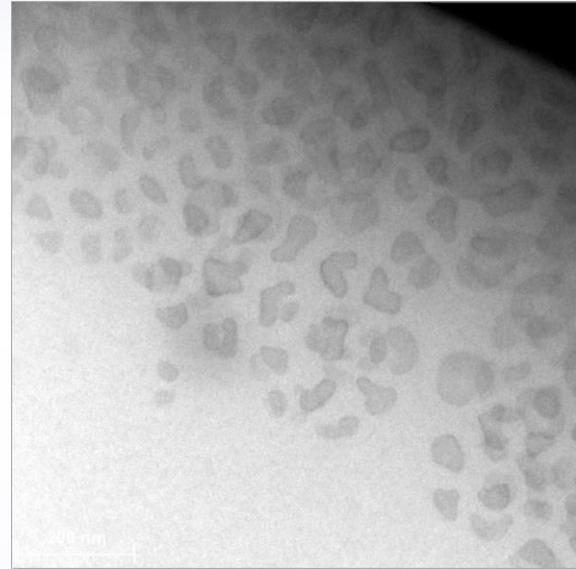
Liposomes
imaged in
flowing aqueous
channel



BSA Crystallization

S. Hoppe

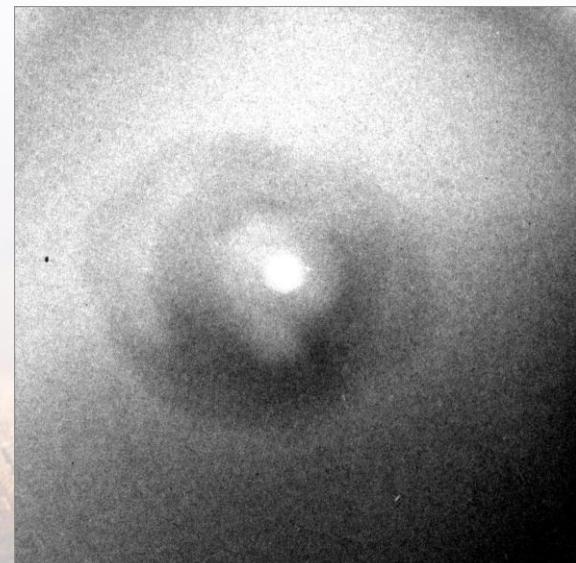
Crystallization of excess
Bovine Serum Albumen
during flow



La Structure Formation

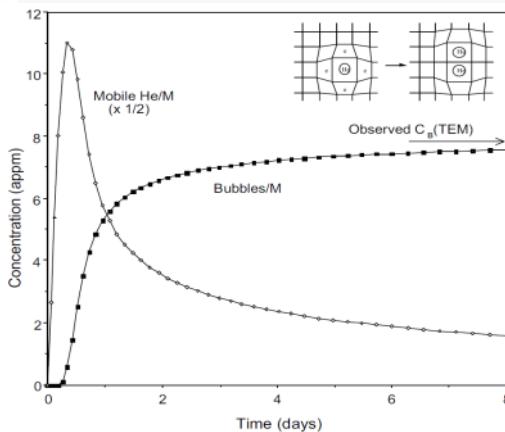
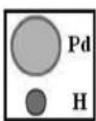
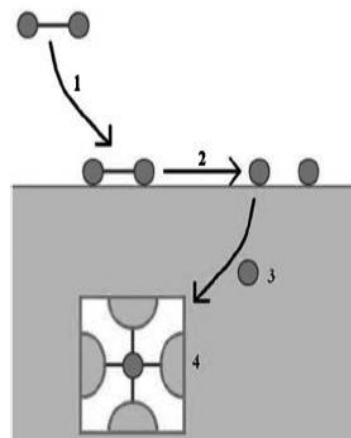
S. Hoppe,
T. Nenoff

La
Nanostructure
form from LaCl_3
 H_2O in wet cell
due to beam
effects



Can *In situ* TEM Address Hydrogen Storage Concerns in Extreme Environments?

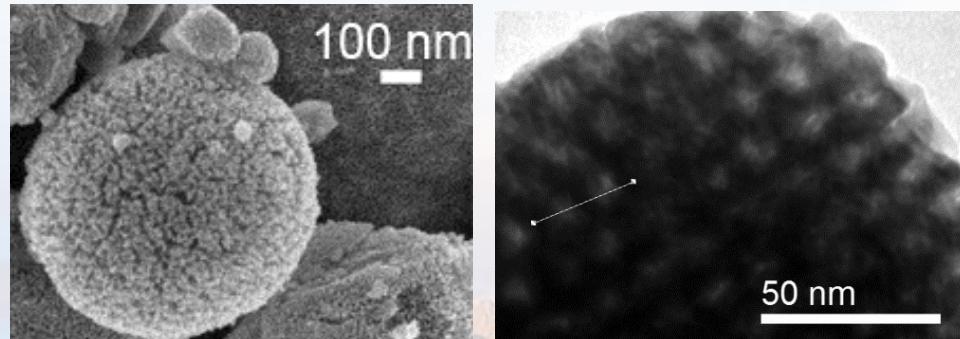
Contributors: B.G. Clark, P.J. Cappillino, B.W. Jacobs, M.A. Hekmaty, D.B. Robinson, L.R. Parent, I. Arslan, & Protochips, Inc.



R. Delmelle, J., Phys. Chem. Chem. Phys. (2011) p.11412

Cowgill, D., Fusion Sci. & Tech., 28 (2005) p. 539
Trinkaus, H. et al., JNM (2003) p. 229
Thiebaut, S. et al. JNM (2000) p. 217

Harmful effects may be mitigated in nanoporous Pd

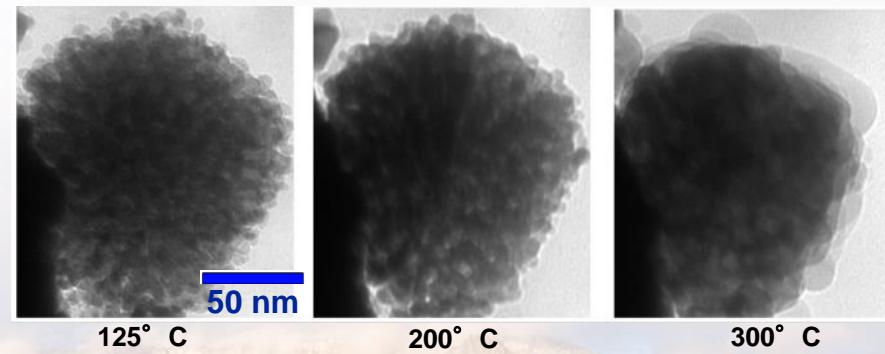


Vapor-Phase Heating TEM Stage

- Compatible with a range of gases
- *In situ* resistive heating
- Continuous observation of the reaction channel
- Chamber dimensions are controllable
- Compatible with MS and other analytical tools



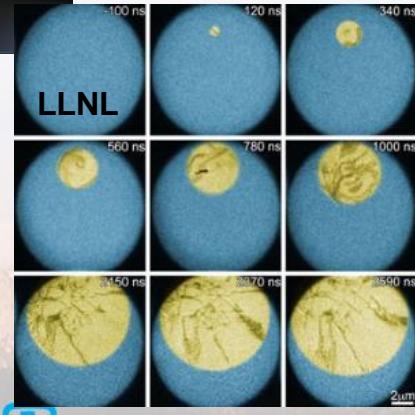
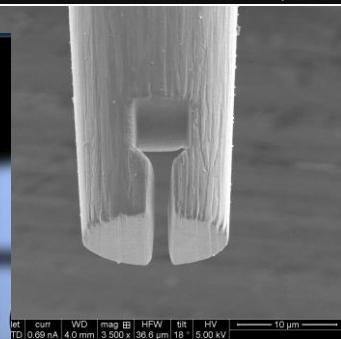
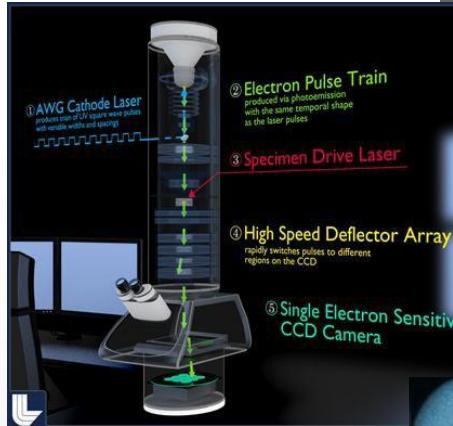
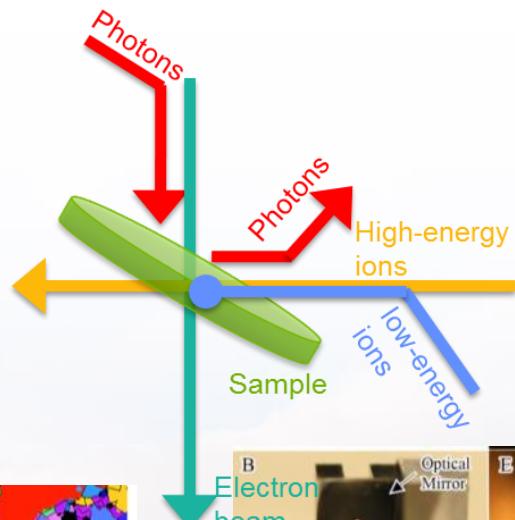
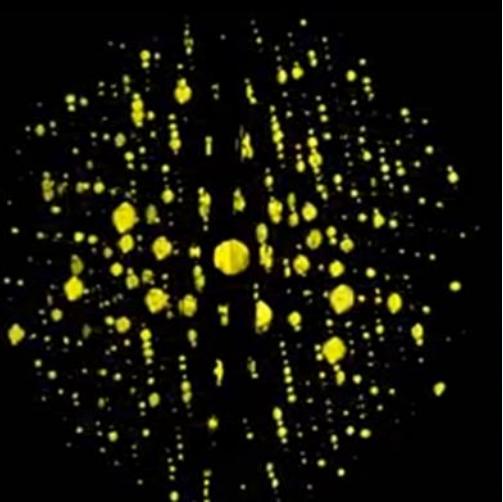
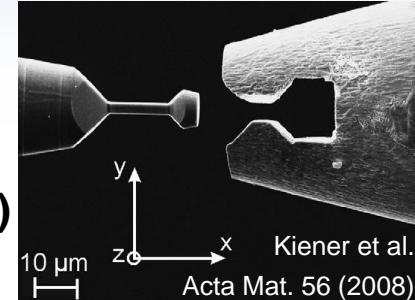
- 1 atm H₂ after several pulses to specified temp.



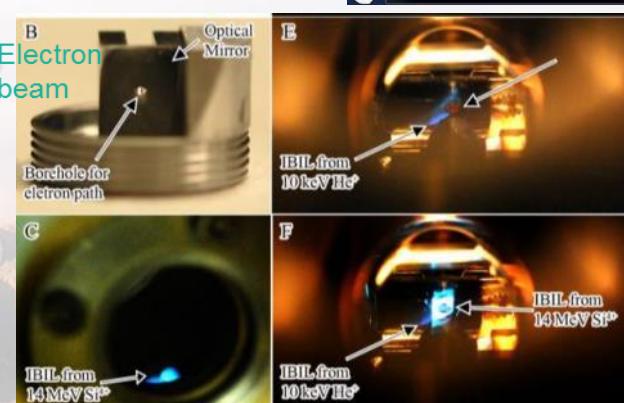
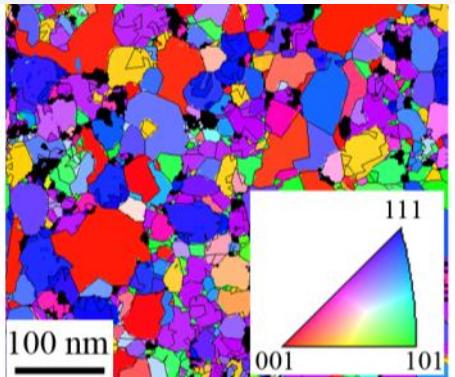
New *In situ* atmospheric heating experiments provide great insight into nanoporous Pd stability

Future Directions Under Pursuit

1. In-situ TEM CL, IBIL (currently capable)
2. *In situ* ion irradiation TEM in liquid or gas (currently capable)
3. PED: Local texture characterization (arriving FY15)
4. Quantitative in-situ tensile/creep experiments (Sample in development)
5. DTEM: Nanosecond resolution (laser optics needed)



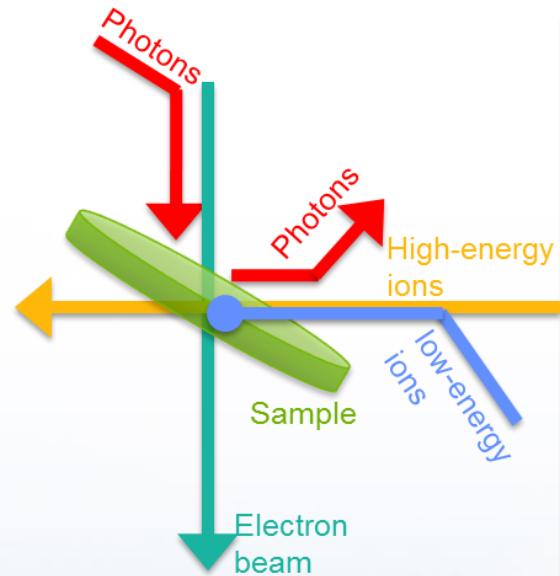
AppFive
NanoMegs



Summary



I³TEM can provide fundamental understanding to key mechanisms in a variety of extreme conditions



Sandia's I³TEM is one of a few in the world

- *In situ* irradiation from H to Au
- *In situ* gas implantation
- Combinations of in-situ techniques

The I³TEM capability are still being expanded...

We are still a long way away from a complete design process that goes from fundamental physics to system engineering

Collaborators:

- IBL: **D.C. Bufford, D. Buller, C. Chisholm, B.G. Clark, B.L. Doyle, S. H. Pratt, & M.T. Marshall**
- Sandia: **B. Boyce, T.J. Boyle, P.J. Cappillino, J.A. Scott, B.W. Jacobs, M.A. Hekmaty, D.B. Robinson, E. Carnes, J. Brinker, D. Sasaki, J.A. Sharon, T. Nenoff, W.M. Mook**
- External: **A. Minor, L.R. Parent, I. Arslan, H. Bei, E.P. George, P. Hosemann, D. Gross, J. Kacher, & I.M. Robertson**

