

*Exceptional service in the national interest*



# Deconvolution of Complex Environmental Effects Active in Nuclear Reactor Materials Through In- situ Ion Irradiation

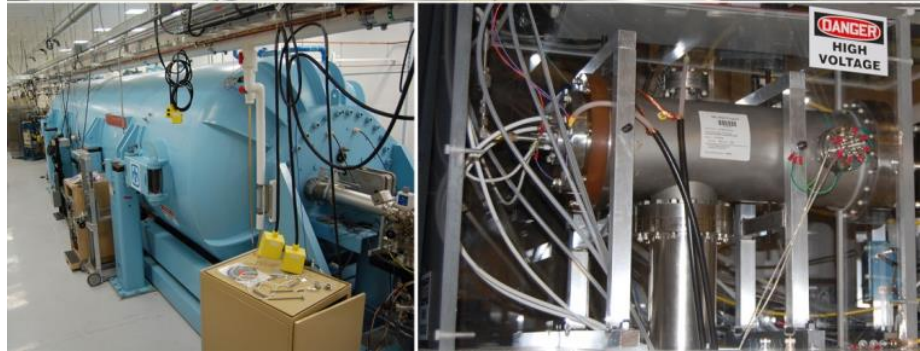
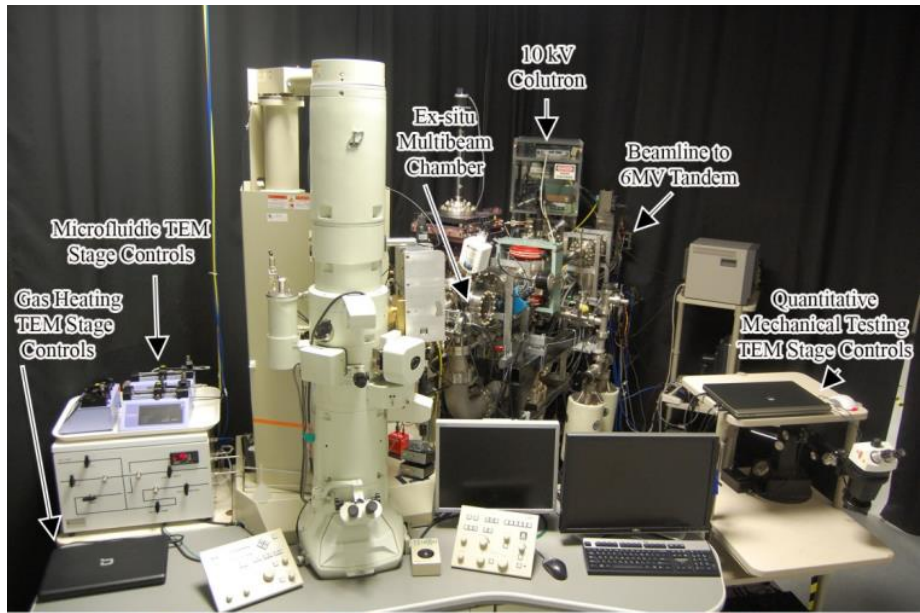
Caitlin Taylor, Christopher Barr, Samuel Briggs, Brittany  
Muntifering, and Khalid Hattar

Sandia National Laboratories

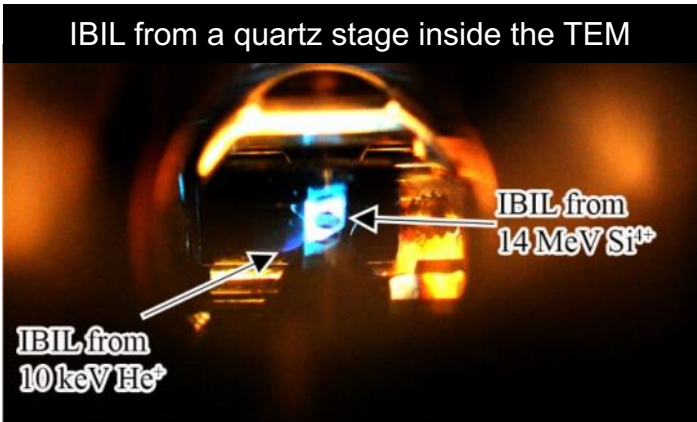


# Sandia's Concurrent In-situ Ion Irradiation TEM Facility (I3TEM)

JEOL 2100 LaB6 microscope

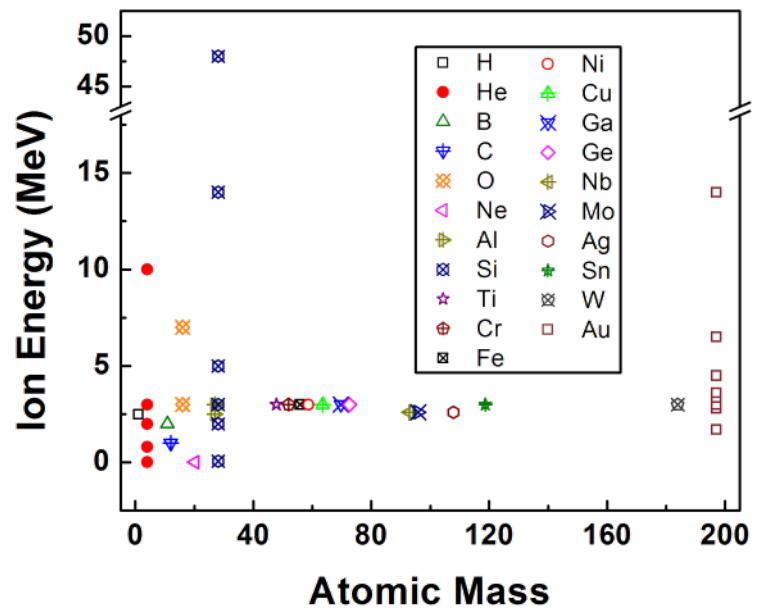


10 kV Colutron – 200 kV TEM - 6 MV Tandem



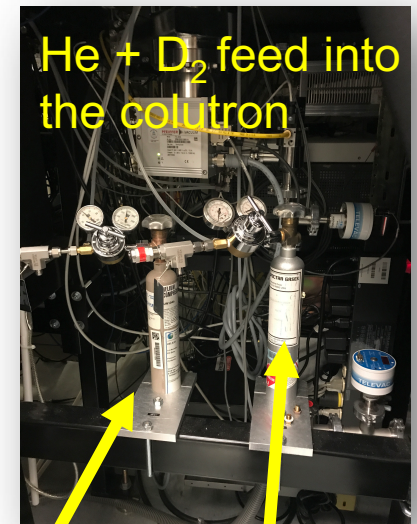
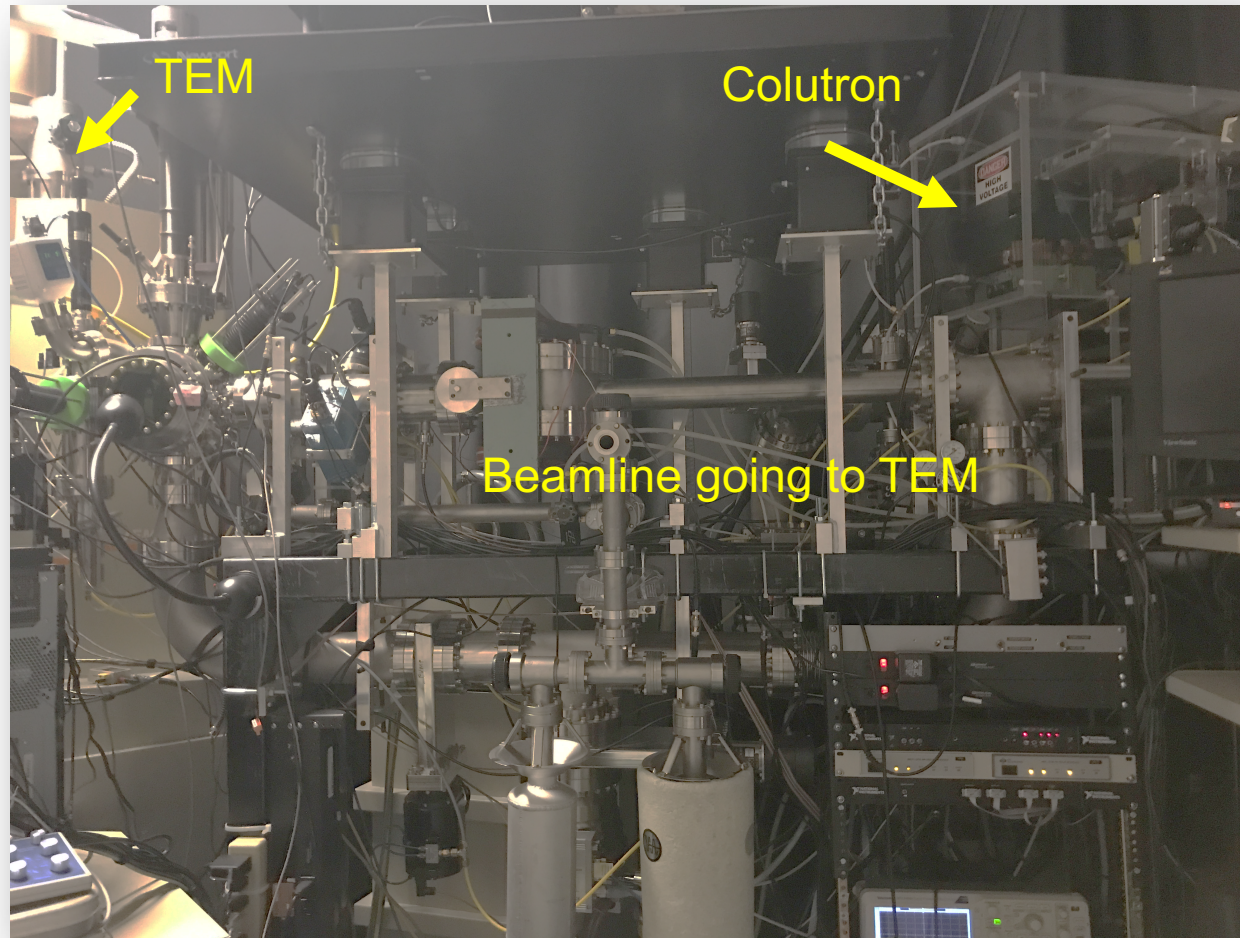
Direct real time observation of ion irradiation, ion implantation, or both with nanometer resolution

Ion species & energy introduced into the TEM





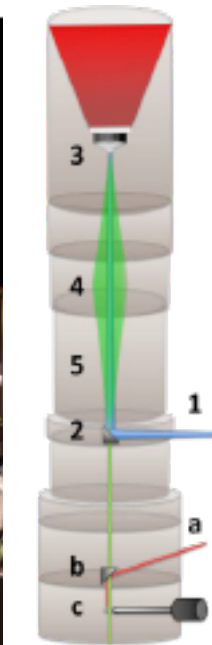
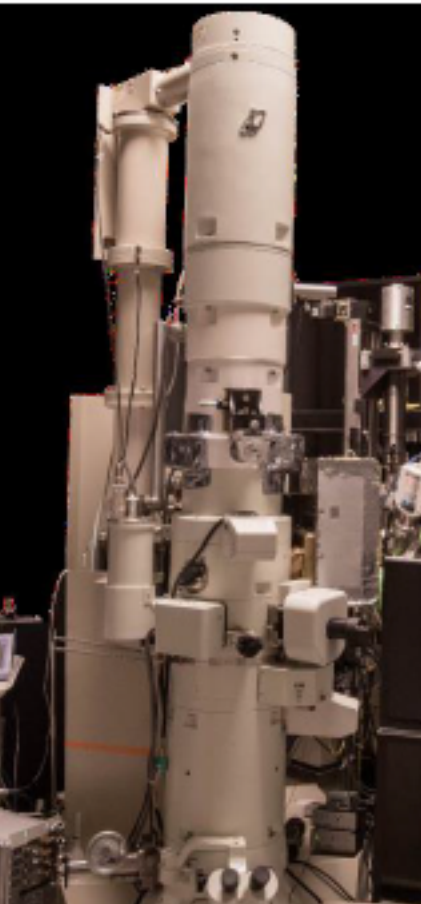
# Sandia's I3TEM (side view of incoming beams)



He bottle      D<sub>2</sub> bottle

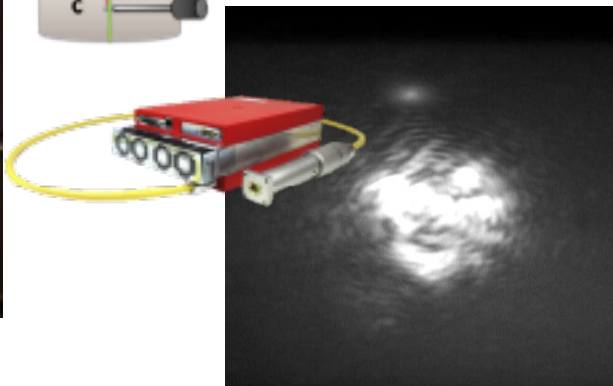
- JEOL 2100 TEM. Samples are tilted to +30° in x to face the ion beam
- Both He and D ions are part of the same ion beam (He + D<sub>2</sub>)

## Unique setups feasible when coupled with IR sample heating:



- Adjustable power 1064 nm infrared specimen (IR) drive laser
- IR laser is reflected directly onto the specimen with metal mirror
- Heat specimens in in situ holders, which otherwise would not be possible

- Laser capabilities:
  - 2-20 Watts
  - Pulsed or continuous operation
  - Tunable spot size (typically 50  $\mu\text{m}$  diameter spot size) – smaller/large possible
  - Positioning mirror, which can be used during laser operation





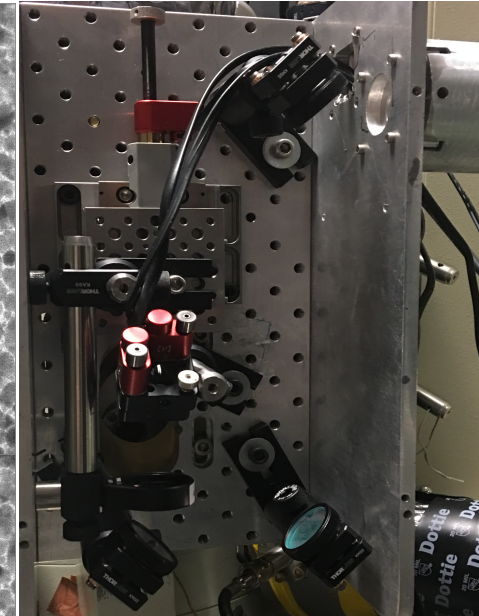
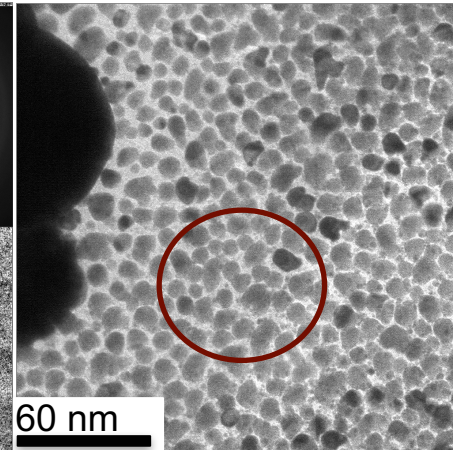
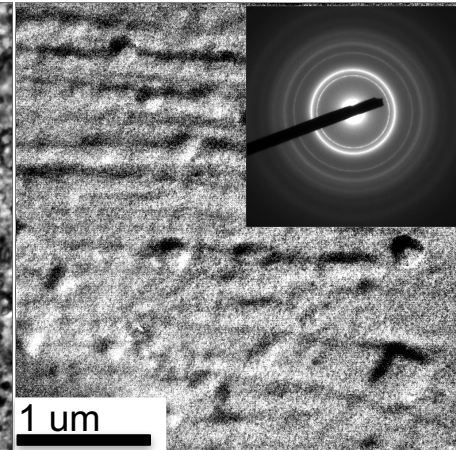
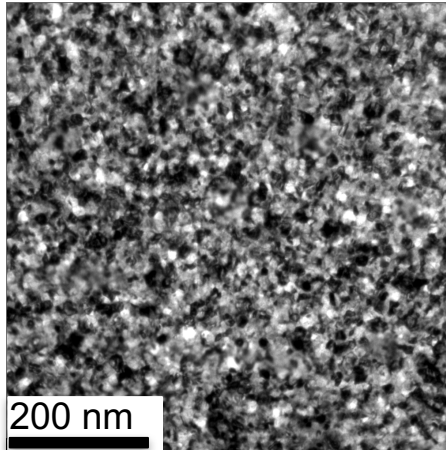
# Synergistic In-Situ Capabilities at I3TEM: Heating via Laser Operation

Pt Grain Growth

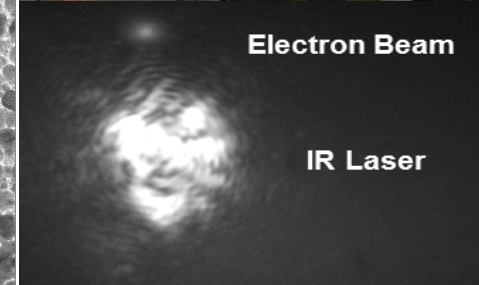
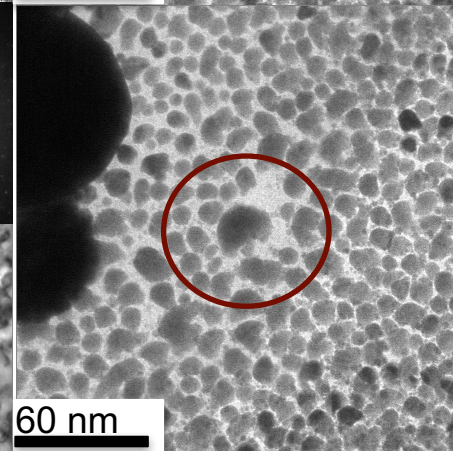
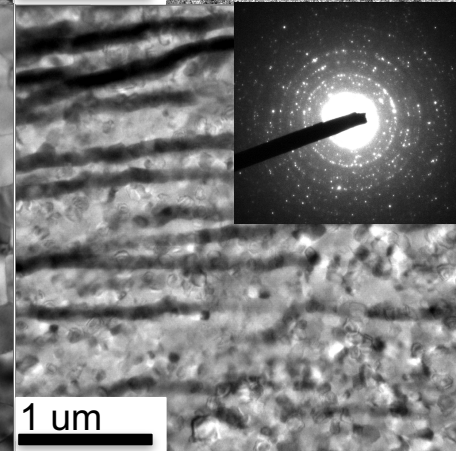
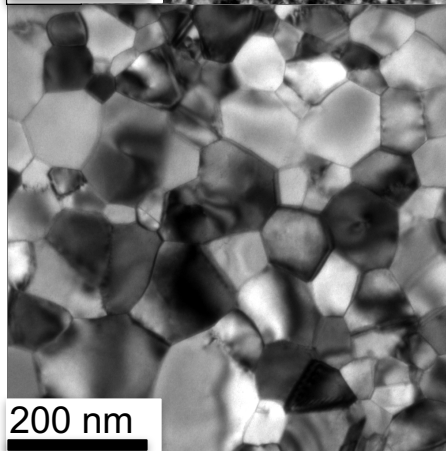
Reactive Multilayer Films

Nanoparticle Sintering (20 nm Au)

Before



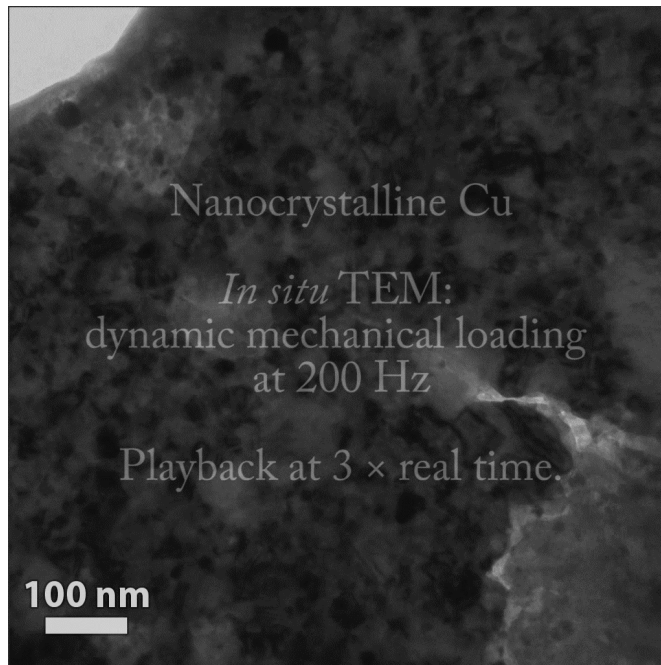
After



- We can now introduce rapid thermal heating with any TEM stage or ion beam conditions
  - Methods underway for accurate temperature determination

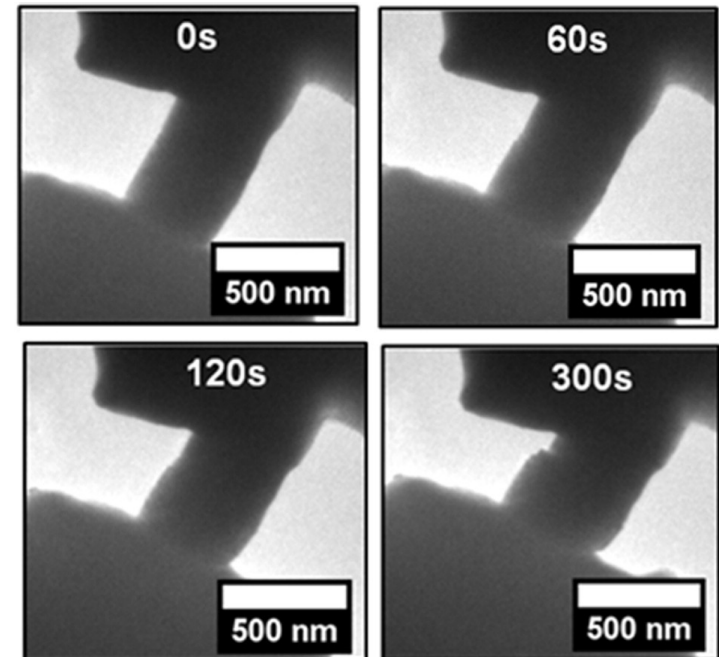
# Synergistic In-Situ Capabilities at I3TEM: Mechanical Testing

- Quantified loading/fatigue/creep



D. Bufford et al., Nanoletters (2016)

- Irradiation induced creep



G.S Jawaharram et al., Scripta Mat **148** (2018) 1-4

## Stage options:

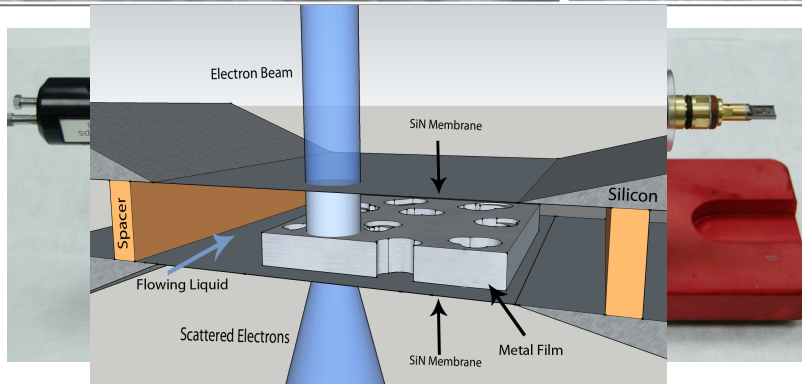
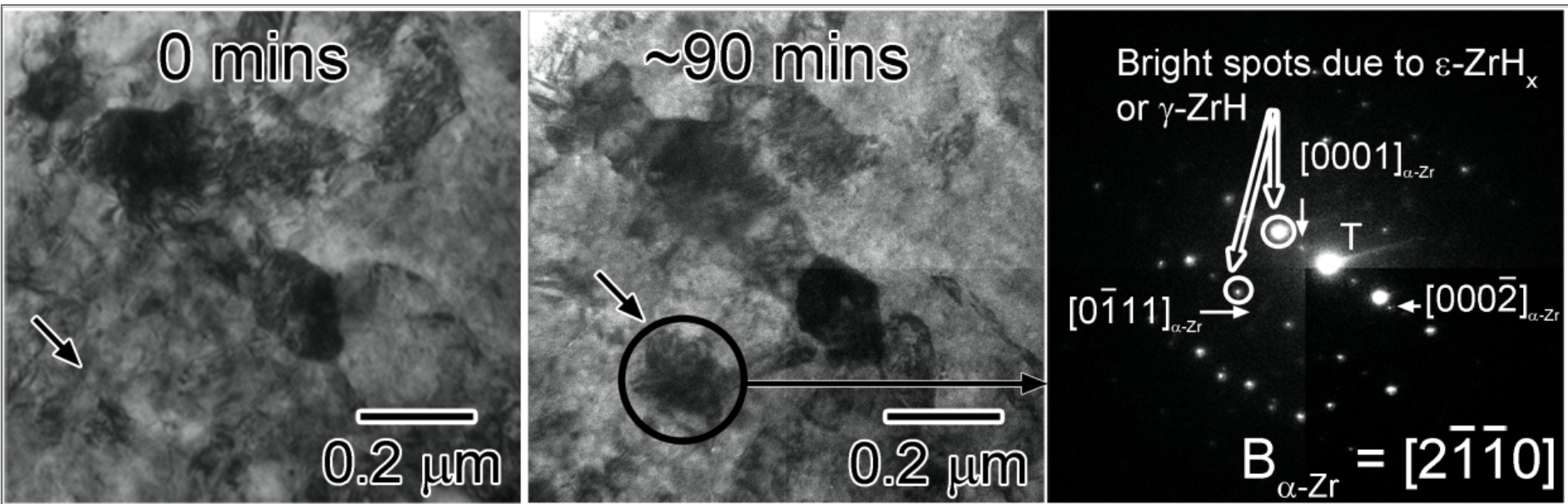
Hysitron PI-95 or Gatan heating/strain holder coupled with IR laser heating





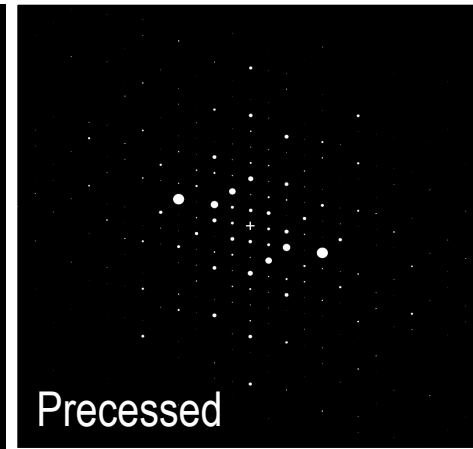
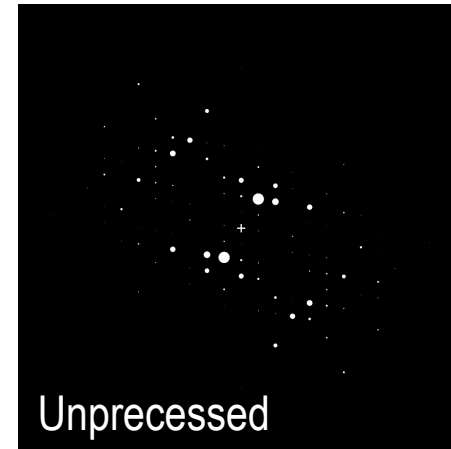
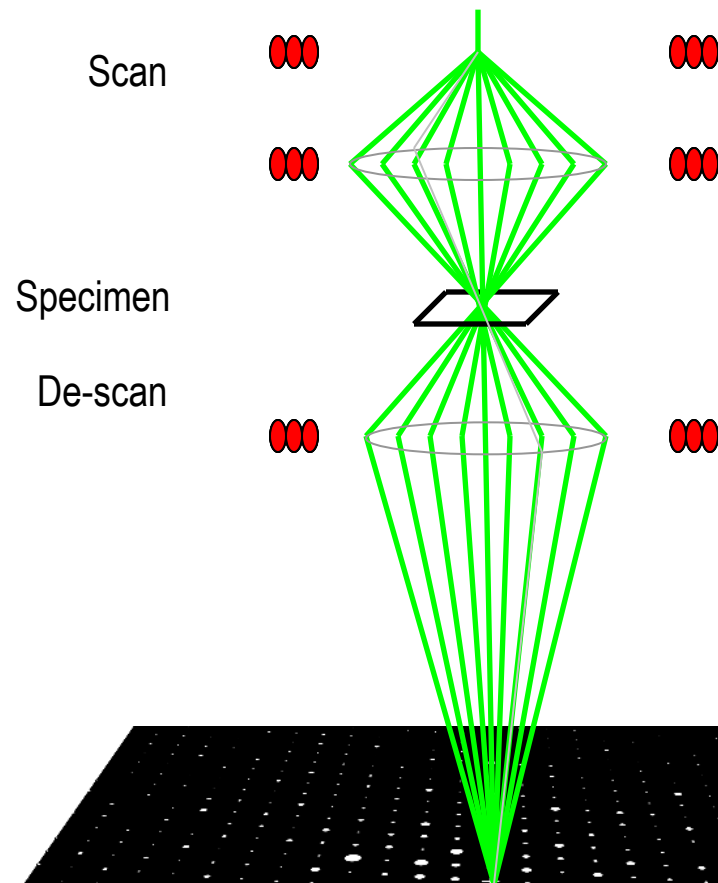
# Synergistic *In-Situ* Capabilities at I3TEM: Corrosion in Zircaloy-2

**Absolute hydrogen pressure:** 327 torr ( $\sim 0.5$  atm),  
**Ramp rate:**  $1^\circ\text{C/s}$ , **Final temperature:**  $\sim 400^\circ\text{C}$ , **Dwell time:**  $\sim 90$  mins



**Hydride formation shown in TEM gas-cell stage at elevated temperature and hydrogen pressure**

# Synergistic Capabilities at I3TEM: Precession Electron Diffraction Microscopy (PED)



## Advantages:

- < 8 nm spatial resolution (LaB6)
- **Near kinematical electron diffraction**
- Symmetry ambiguities are resolved
- Fast and automated acquisition
  - $\approx 200$  grains in 15 min.
  - Angular resolution  $\approx 1^\circ$

(Diffracted  
amplitudes)



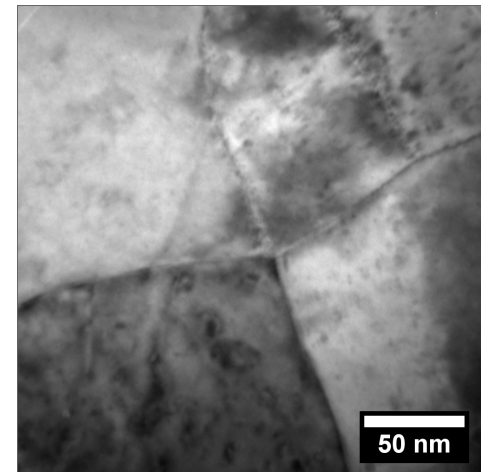
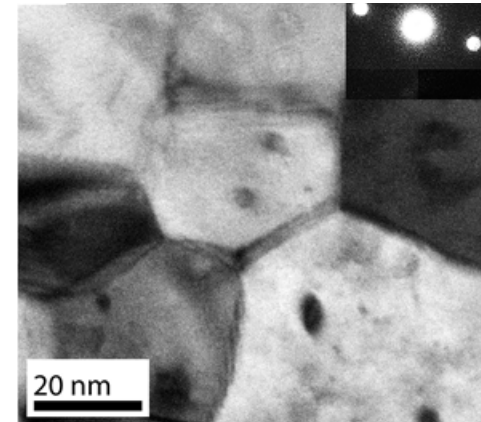
**NanoMEGAS**  
Advanced Tools for electron diffraction

# Two Case Studies of Irradiation Type Experiments Feasible at I3TEM

## Most Common In-situ TEM Irradiation Setups

- (1) Heavy ion irradiation into NC Pt
- (2) Helium implantation and annealing in NC Pd – “Briefly”

All experiments (ion irradiation, helium implantation, and triple combined beam) can be coupled with other in-situ techniques previously detailed for additional complexity and simulated environments



## **Case 1: Grain Size Effects Under Irradiation in NC Pt**

Project collaborators: Nan Li and Brad Boyce



# NC Grain Size Effects: Motivation

- **Increase GB density** → numerous nanocrystalline studies indicating reduced defect size and/or density of defect (dislocations and voids) with NC grains
- Defect size and density reduction indicate improvements (reduction) in defect size and density from coarse grain counterparts
- **Very limited exploration within the nanocrystalline regime**

Can we use in-situ TEM irradiation coupled with defect quantification to understand if any variation exist within nanocrystalline regime (grain size less 100 nm)?

NC Pd defect trends at RT

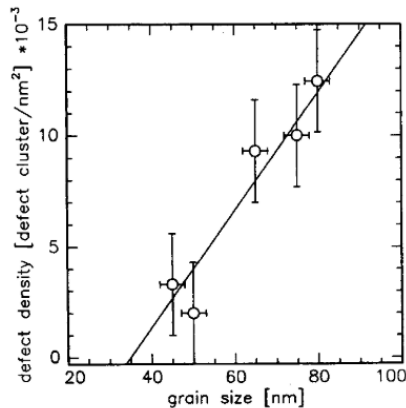
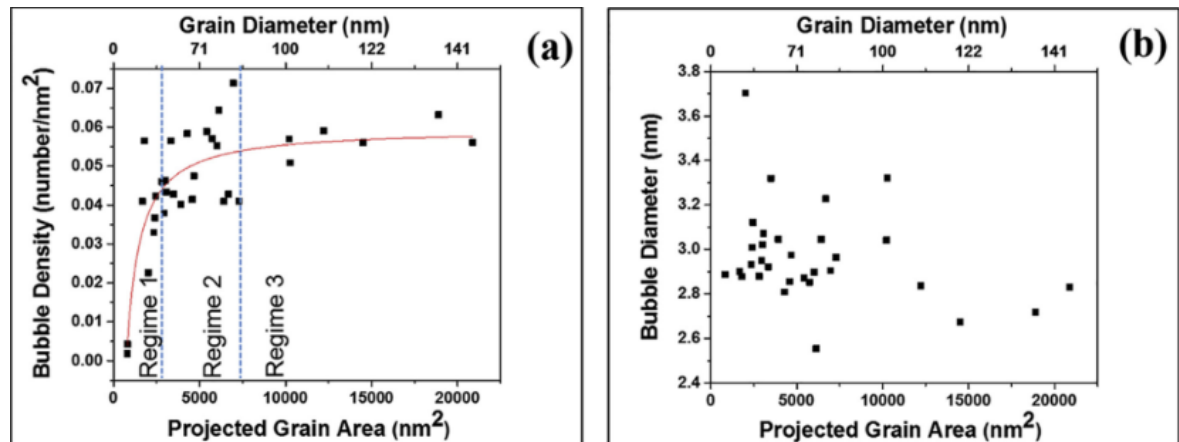


Fig. 8. Defect densities vs. grain size on the irradiated Pd sample (240 keV,  $2 \times 10^{16}$  Kr/cm<sup>2</sup>).

Rose et al. (1997)

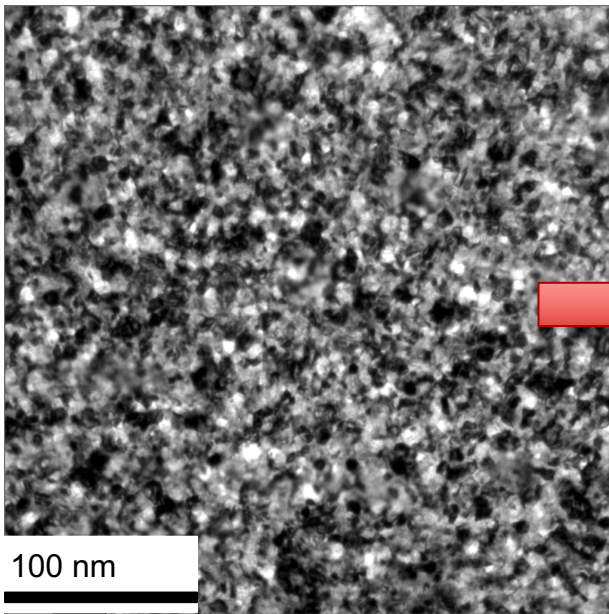
NC Fe Helium bubble trends at 427°C



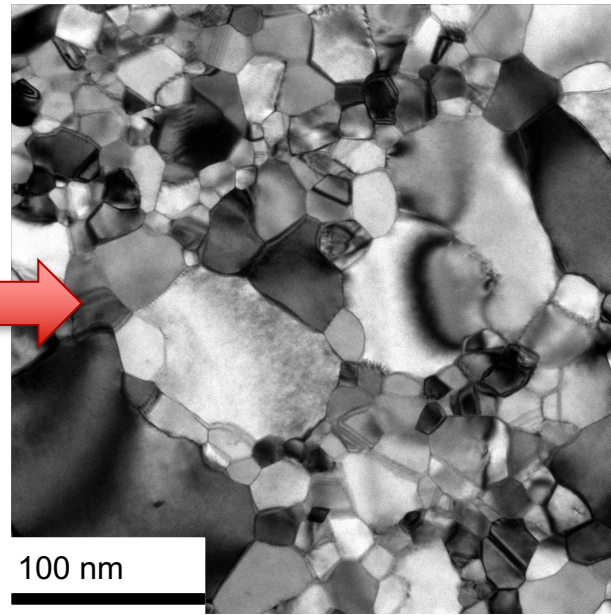
El-Atawani et al. 484 (2017) 236-244.

# NC Grain Size Effects : Methodology

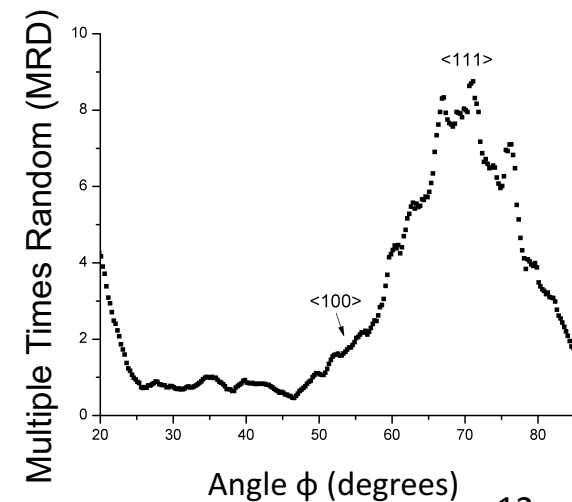
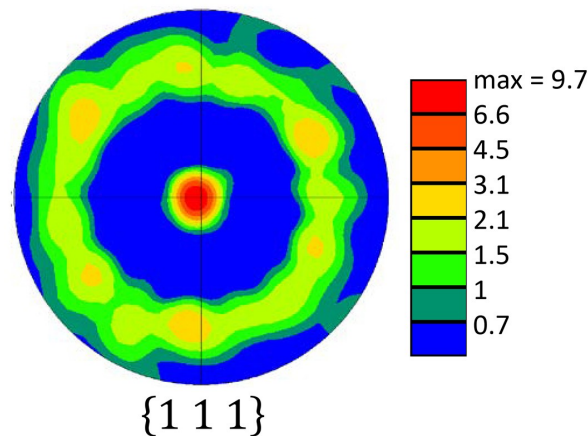
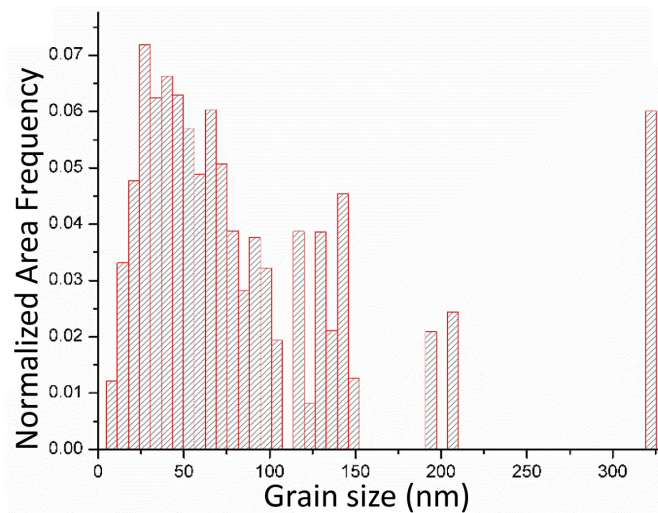
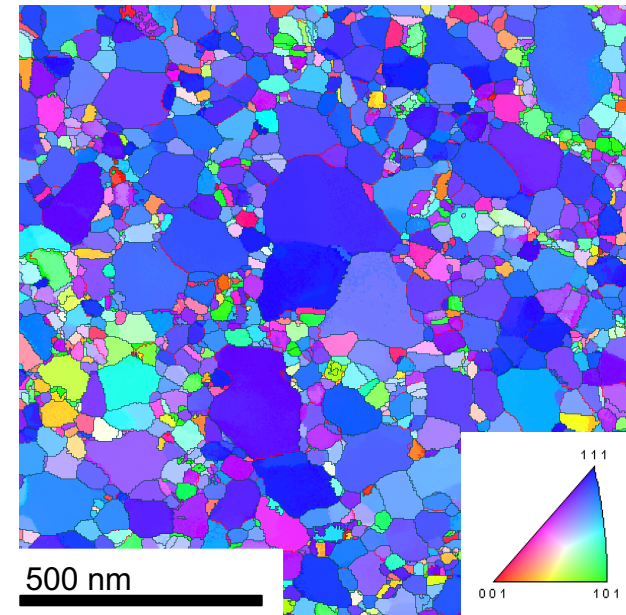
As-deposited NC Pt



Annealed NC Pt (500 ° C/2hrs)

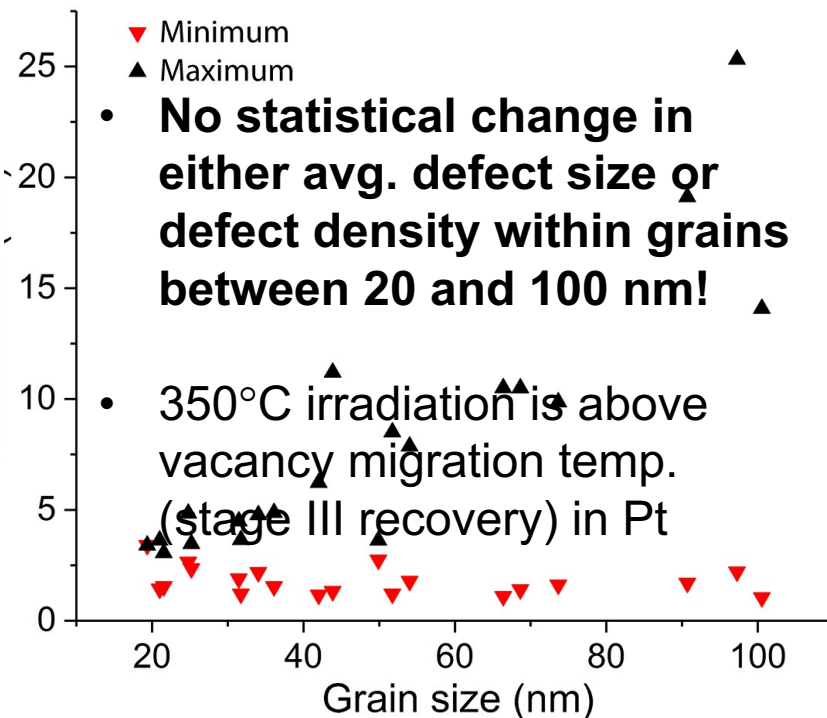
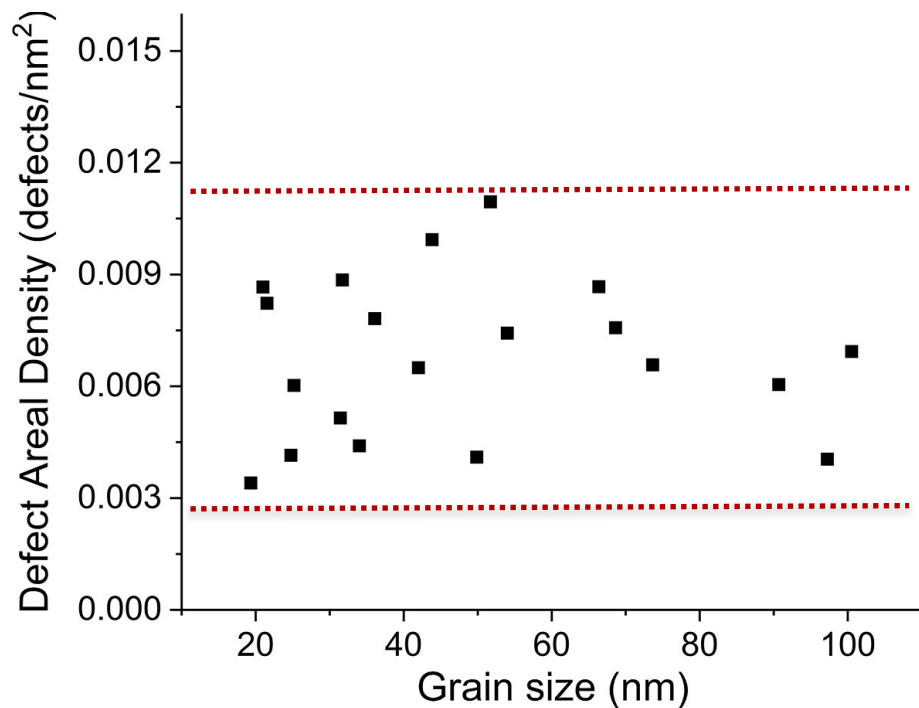


Annealed PED Map



# NC Grain Size Effects : Irradiation and Defect Trends

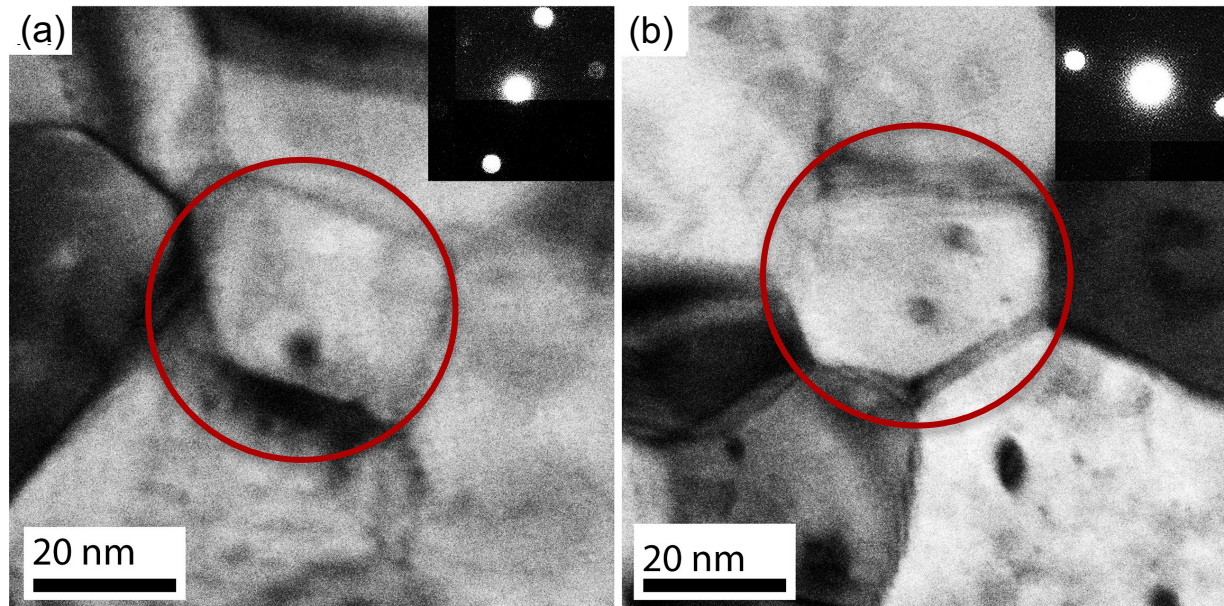
- In-situ TEM irradiations: 2.8 MeV Au<sup>4+</sup> at a flux of  $4.7 \times 10^{10}$  ions/cm<sup>2</sup>s to a fluence of  $1.7 \times 10^{14} \frac{\text{ions}}{\text{cm}^2}$  at 350°C using a Gatan double tilt heating stage
- **Average final dpa  $\approx$  3 dpa**
- NC grains with a slightly deviated  $g = \langle 111 \rangle$  diffraction condition using two-beam kinematic bright-field TEM utilizing processed nanobeam electron diffraction



- **No statistical change in either avg. defect size or defect density within grains between 20 and 100 nm!**
- 350°C irradiation is above vacancy migration temp. (stage III recovery) in Pt



# NC Grain Size Effects: Cautionary Tale in Defect Counting

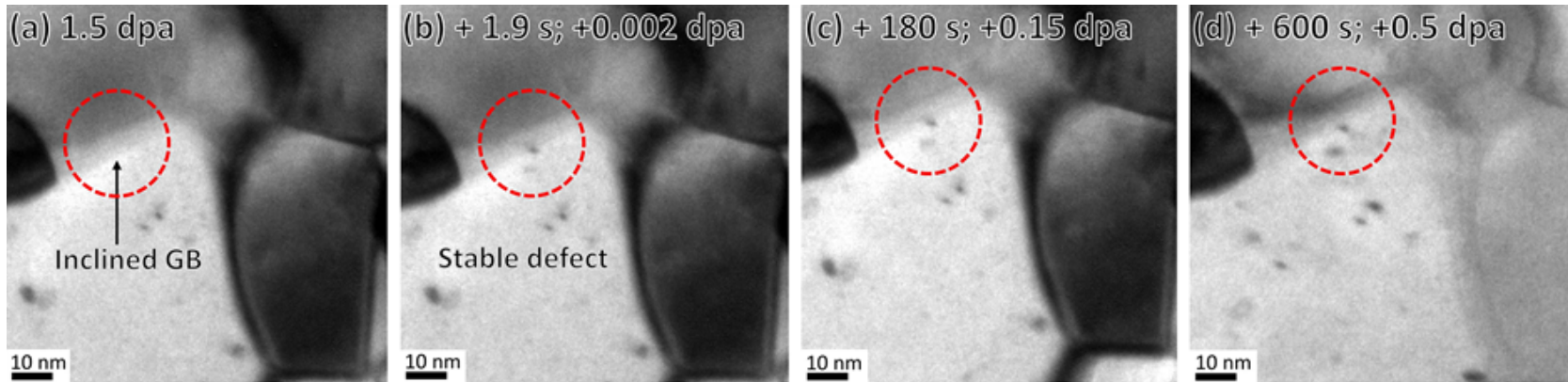


- Two grains  $\rightarrow$  20 nm in size (measured via mean intercept length)
- **How do you count the defects in these grains? 1 to 3 defects nearly full range of density!**
  - Grain A = 1 defect  $\rightarrow 3.3 \times 10^{-3}$  defects/nm<sup>2</sup>
  - Grain B = 3 defects  $\rightarrow 8.6 \times 10^{-3}$  defects/nm<sup>2</sup>
- **With decreasing grain size, # of defects become both more challenging and critical for any type of accurate defect density**

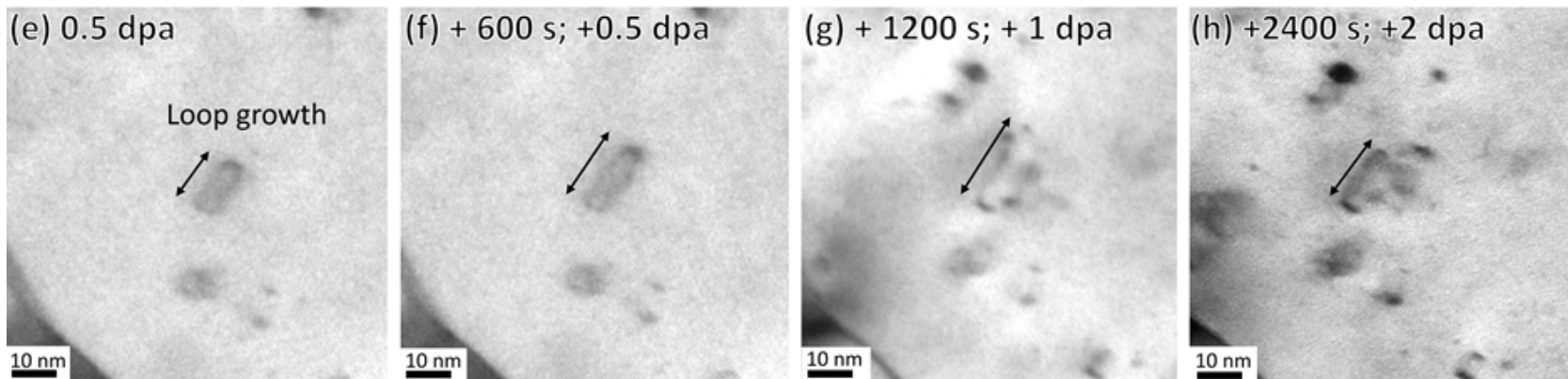


# NC Grain Size Effects: Lack of Defect Density Reduction In-situ Observations

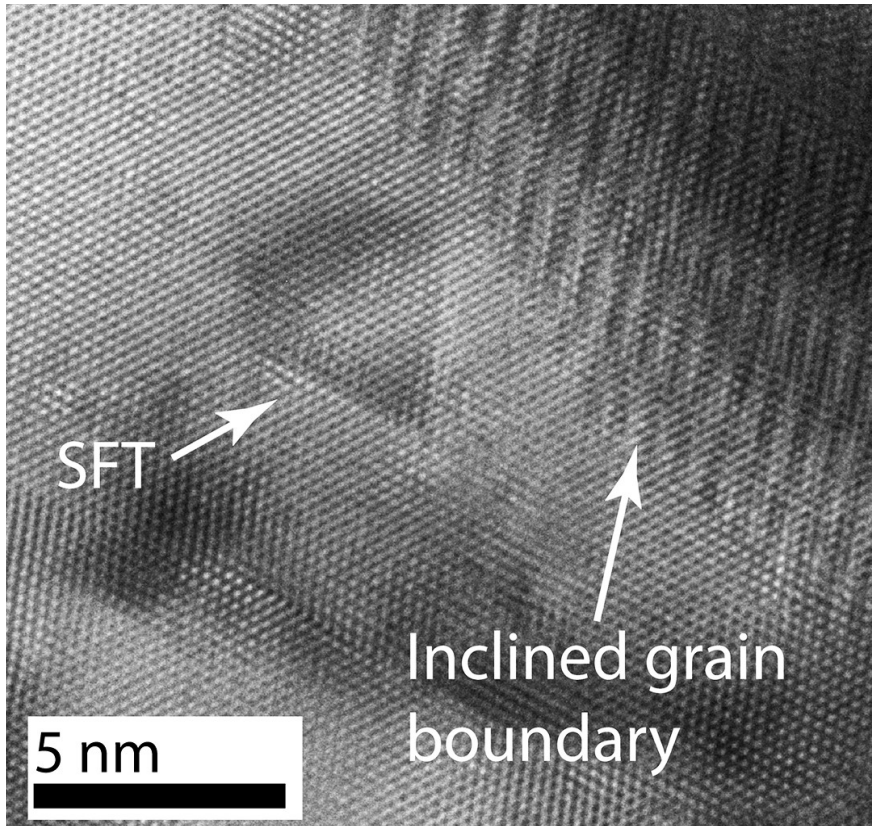
## Defect evolution near GBs:



## Defect evolution in grain interior

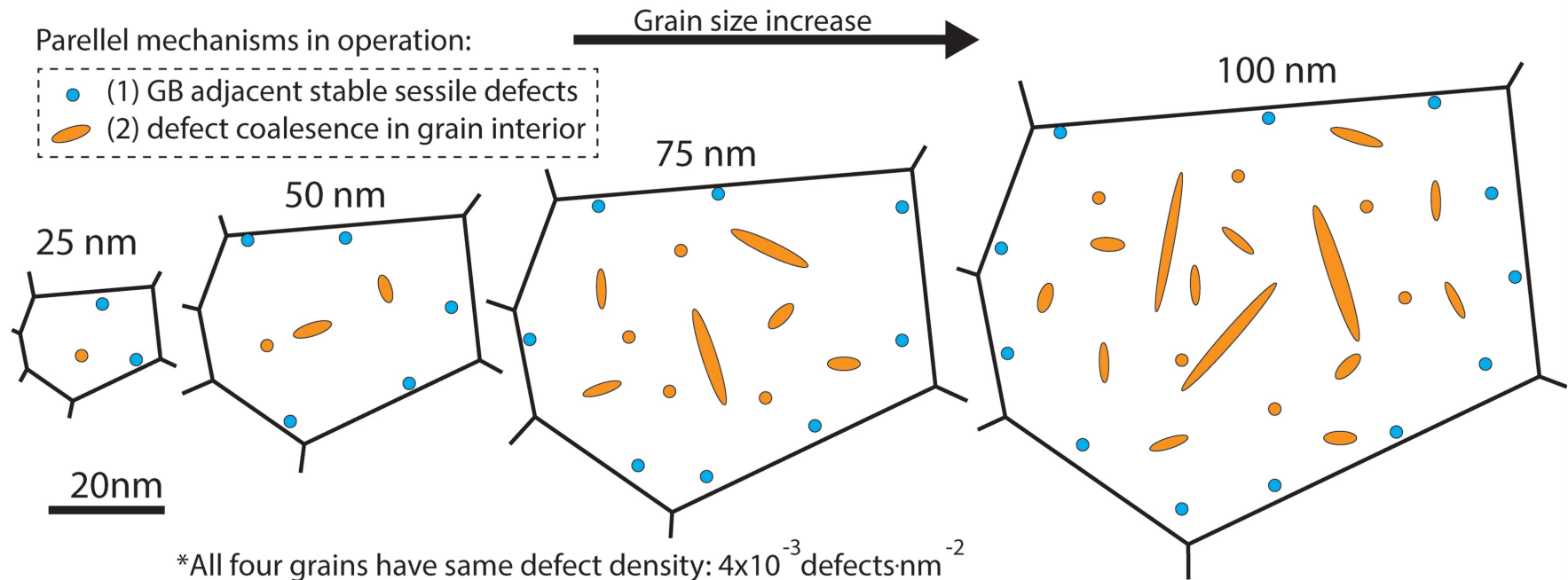


# NC Grain Size Effects: Small loop stability near GBs



- Stable sessile defect clusters (1 to 3 nm) adjacent to most GB
- As previously shown, only a small number of these stable defects can impact (increase defect density in small  $> 40$  nm grains)
- “Small stable” defects  $\rightarrow$  majority assumed to be SFTs

# NC Grain Size Effects: Proposed Mechanism



- We propose a threshold or breakdown of a reduction in irradiation induced defect density in NC Pt → Analogous to the classic Hall-Petch strengthening breakdown within the nanocrystalline regime
- Results observed in thin film Pt highlight a lack of radiation tolerance (as defined as a reduction in defect size and density) between 100 and 20 nm
- **Only one temperature, flux, fluence, material system → more efforts to expand beyond this limited scope!**

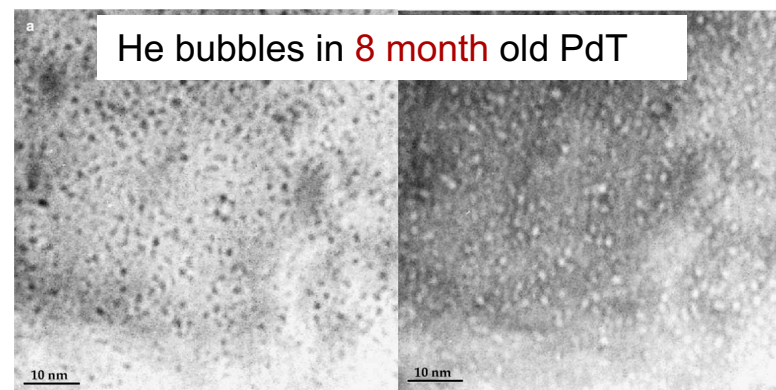
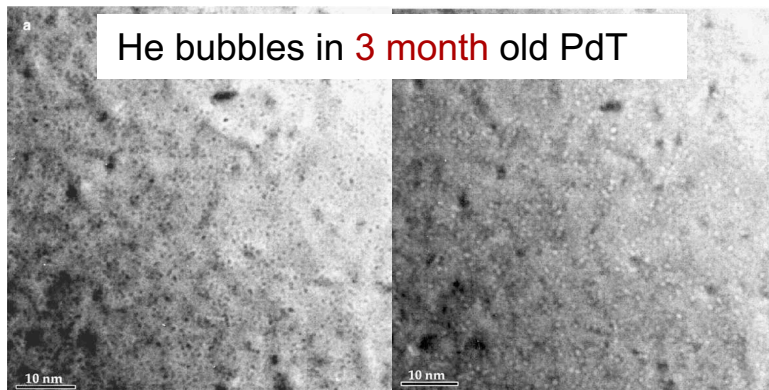
## **Case 2: Helium Bubble Nucleation and Growth in Pd**

Palladium work completed by Caitlin Taylor (SNL),  
Collaboration with Joshua Sugar (SNL) and Dave  
Robinson (SNL)



# Palladium: Considered for solid-state $^3\text{H}$ storage → Result in $^3\text{He}$ bubbles and blistering

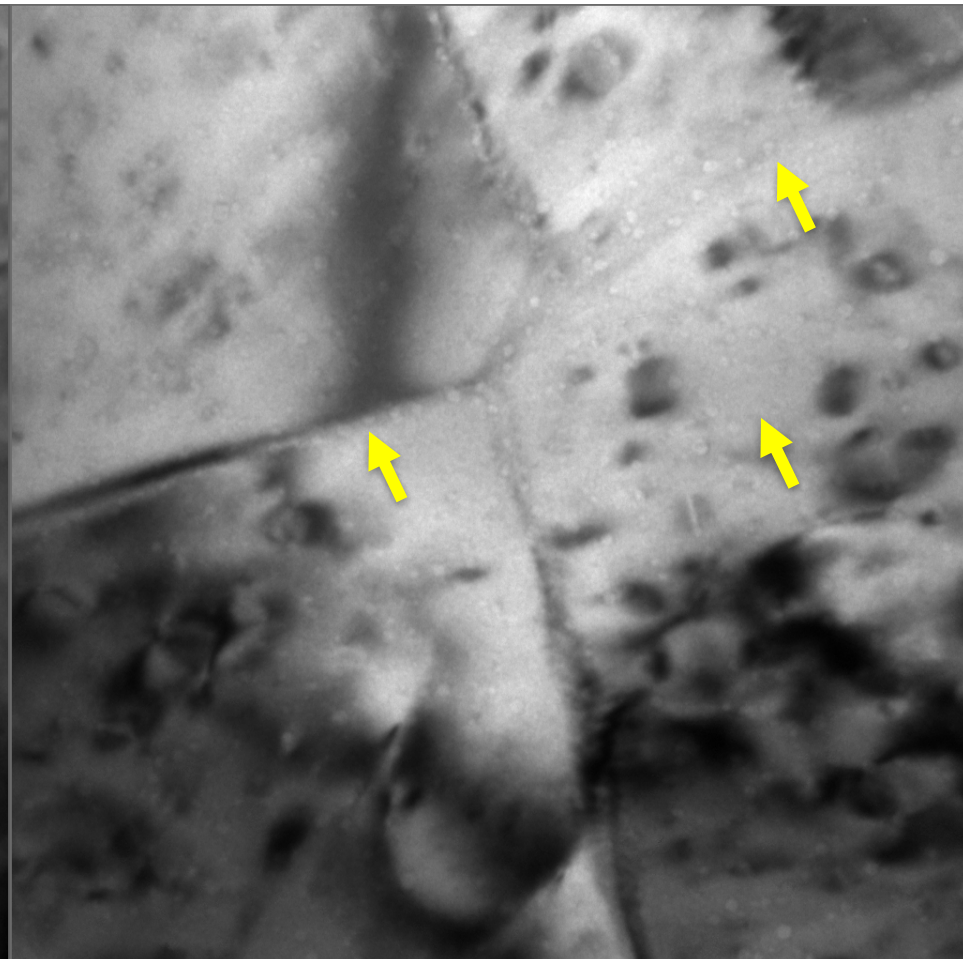
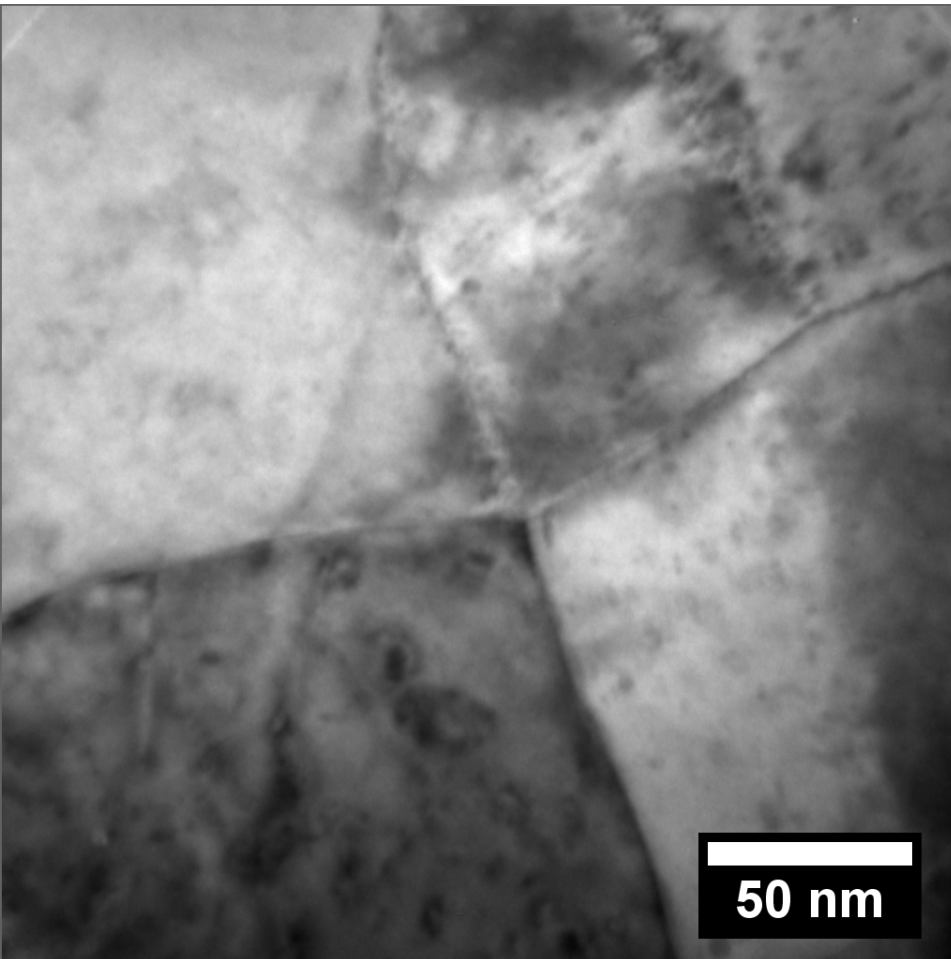
- Palladium is being considered for solid-state tritium storage
- Helium accumulation
  - Due to rapid  $^3\text{He}$  production in tritiated materials, **He bubble formation will eventually lead to blister formation**
  - Blisters bursting the surface release  $^3\text{He}$  and  $^3\text{H}$  from the material.
- Bubble nucleation mechanisms in PdT are currently unknown.  
Necessary for predictive aging models



# Nanometer sized bubbles nucleated inside grains and at boundaries during in-situ 400°C implantation

Before

After

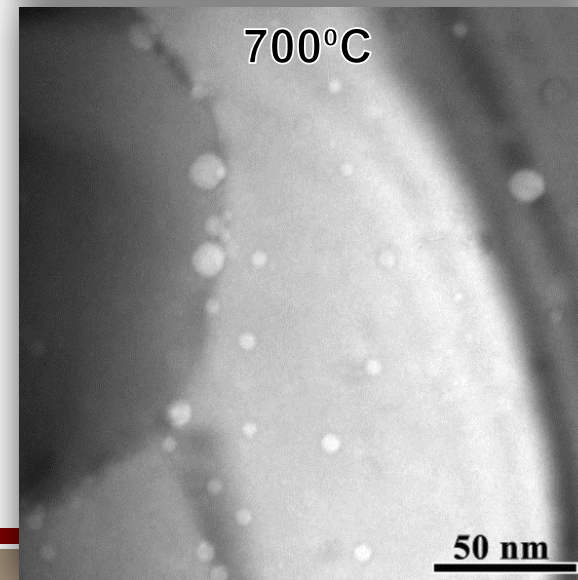
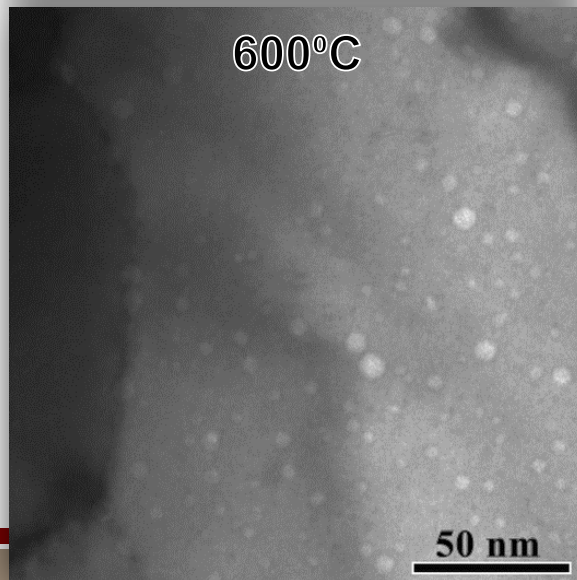
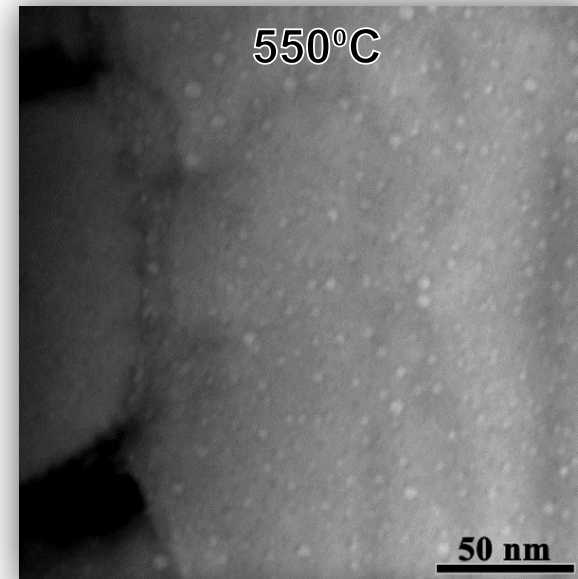
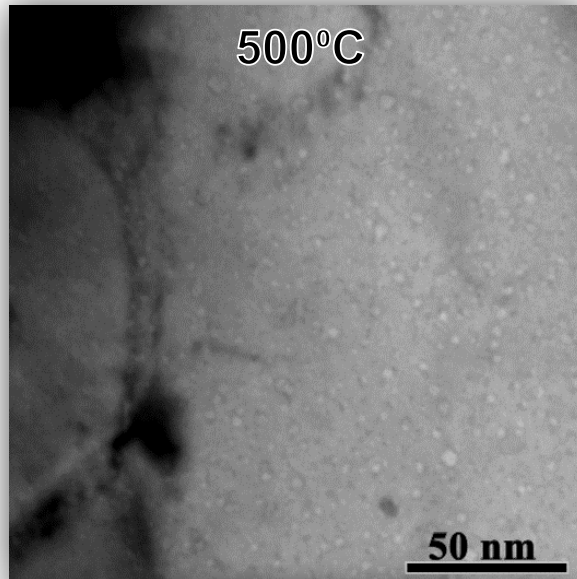


- During in-situ helium implantation, bubbles were observed forming simultaneously at the boundaries and inside the grains. 5 uA beam current, spot 10mm<sup>2</sup>, 2 hrs total implantation
- Bubbles were an average of 1.9 nm in diameter after the implantation.

# Bubble migration first observed at 450°C, with bubble growth first being observed at 500°C

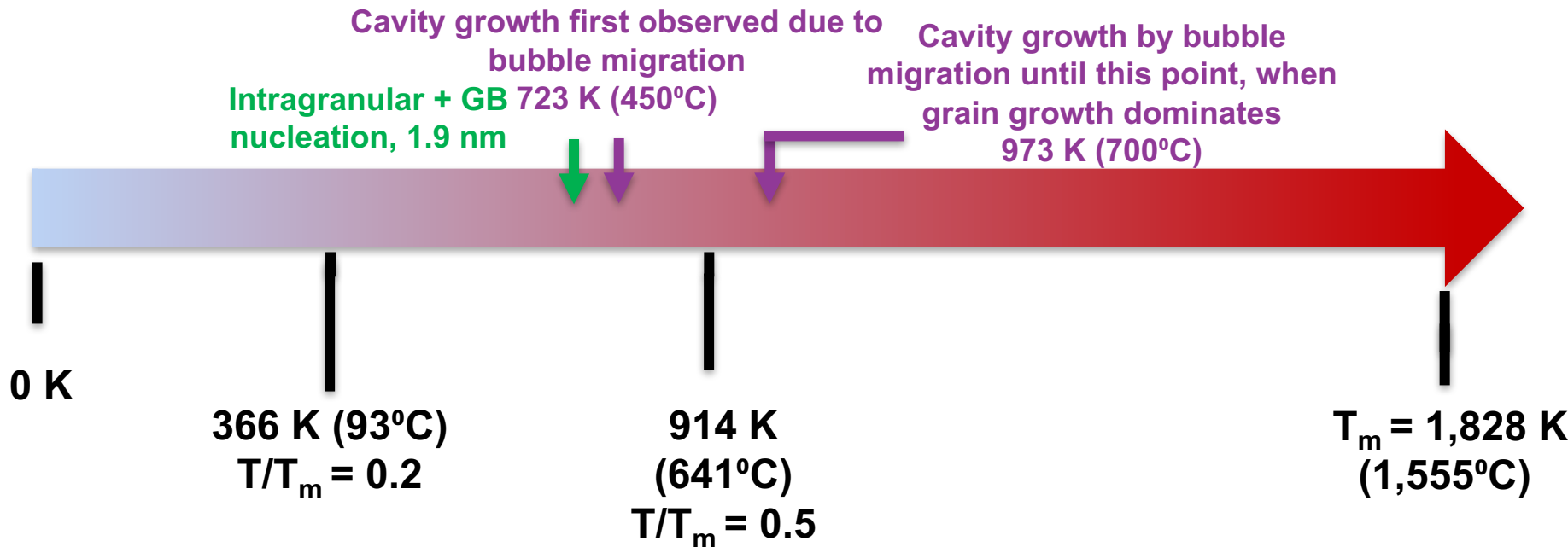
Sample implanted @ 400°C

Video 12x  
speed



# Summary of helium bubble kinetics observed during in-situ implantation and annealing

400°C implant

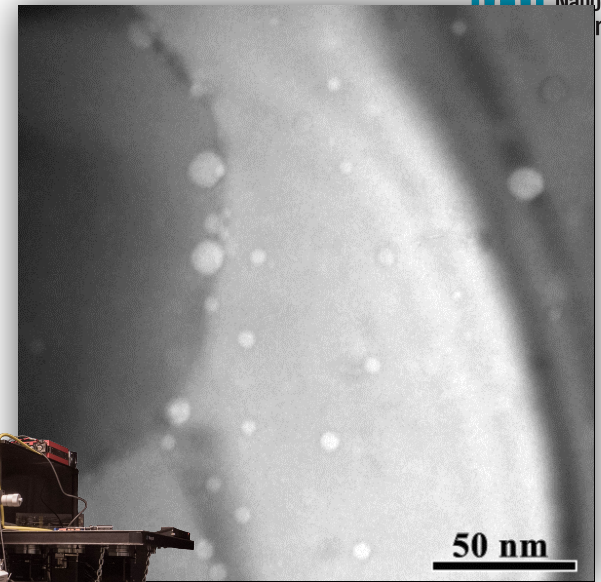
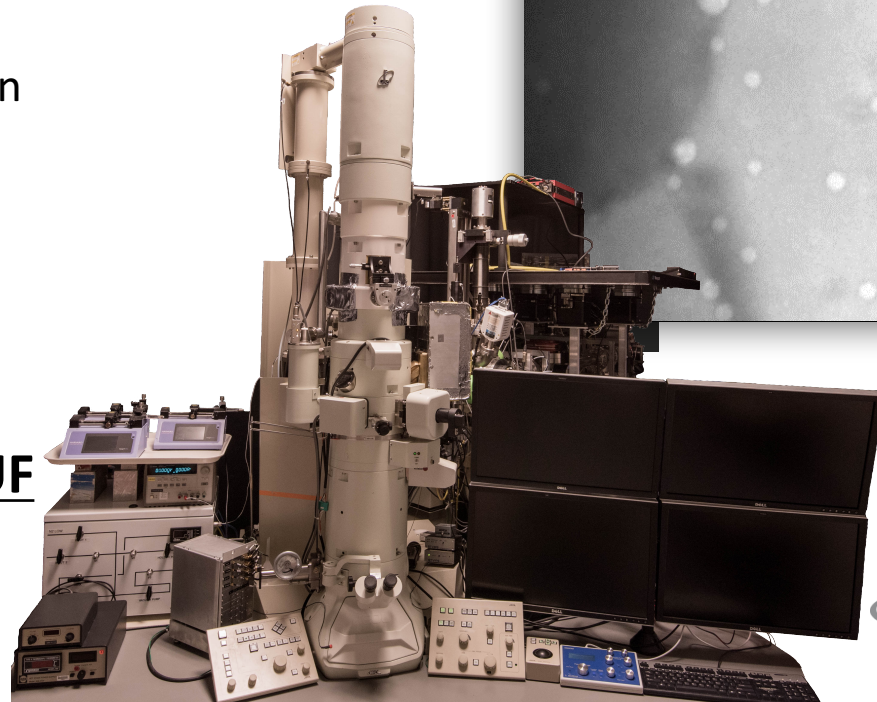


- At 400C, entire bubbles are observed migrating to the GB, promoting cavity growth at GBs.
- Intragranular growth is first observed at this temperature due to He or small He-V cluster diffusion from the lattice.
- At 450C, cavity migration occurs everywhere, eventually cavities become faceted and larger at GBs.



# Summary

- Sandia's I<sup>3</sup>TEM unique CINT and NSUF facility
- Potential for wealth of new *in situ ion irradiation* capabilities
- *In situ* high energy ion irradiation to Au
- > 10 TEM stages with various capabilities
  - Mechanical to Liquid Cell microscopy feasible
- **Facility accessible via NSUF and CINT via user proposals**



Sandia's I<sup>3</sup>TEM provides a wealth of interesting observations across various complex environments

Sandia National Laboratories is a multi-mission laboratory managed and operated by National Technology & Engineering Solutions of Sandia, LLC., a wholly owned subsidiary of Honeywell International, Inc., for the U.S. Department of Energy's National Nuclear Security Administration under contract DE-NA0003525.



U.S. DEPARTMENT OF  
**ENERGY**

Office of  
Science