

Structure and Dynamics in Ion-Conducting Polymers from MD Simulations

SAND2018-2416C

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APS March Meeting
March 9, 2018



Sandia National Laboratories is a multi-mission laboratory managed and operated by National Technology and Engineering Solutions of Sandia, LLC., a wholly owned subsidiary of Honeywell International, Inc., for the U.S. Department of Energy's National Nuclear Security Administration under contract DE-NA-0003525.

Acknowledgments

Sandia National Labs

Jason Koski
Mark Stevens

Joe Cordaro
Dale Huber

Lauren Abbott
Eric Sorte
Cy Fujimoto
Todd Alam

University of Pennsylvania

Ben Paren
Robert Middleton
Phil Griffin
Karen Winey

NIST

Jacob Tarver
Madhu Tyagi
Christopher Soles



Funding

Sandia LDRD
Center for Integrated
Nanotechnologies

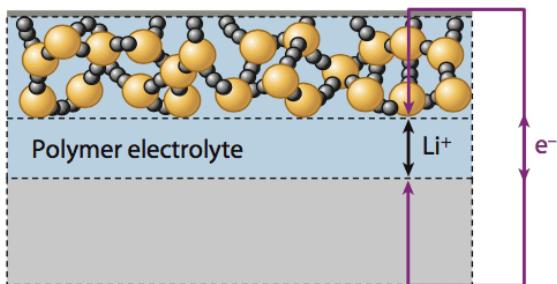


Ion-conducting polymers in energy storage

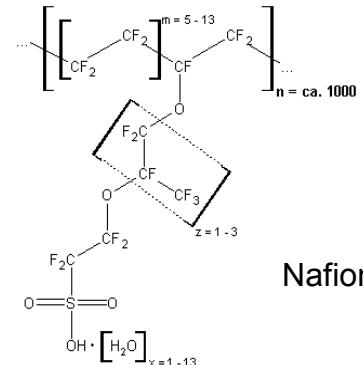
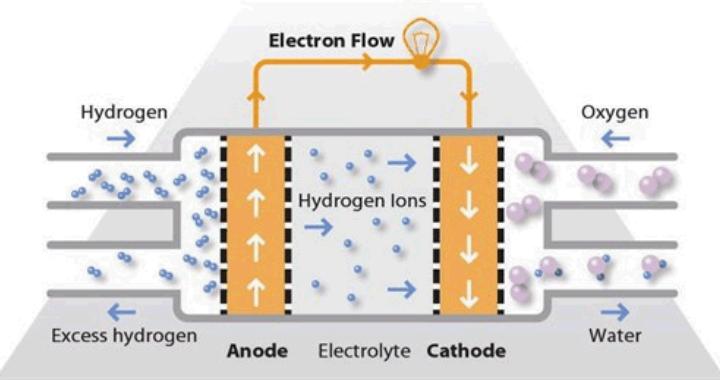
membranes/electrolytes for

- Li ion batteries
- flow batteries
- fuel cells

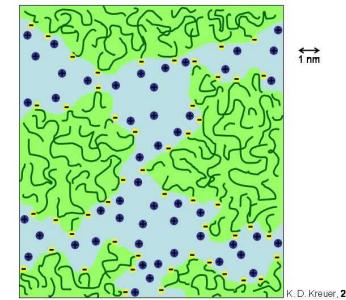
a Positive porous electrode
 $\text{Li}^+ + \text{e}^- + \text{FePO}_4 \leftrightarrow \text{LiFePO}_4$



Negative electrode
 $\text{Li} \leftrightarrow \text{Li}^+ + \text{e}^-$



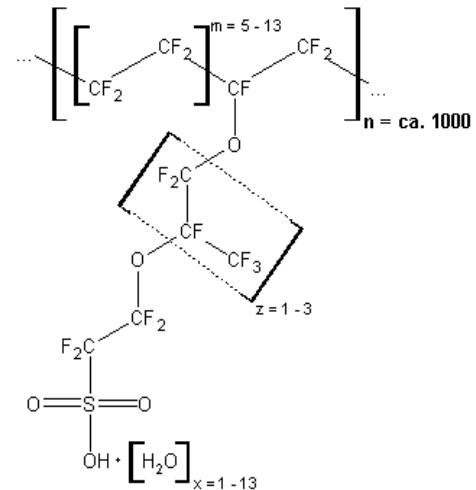
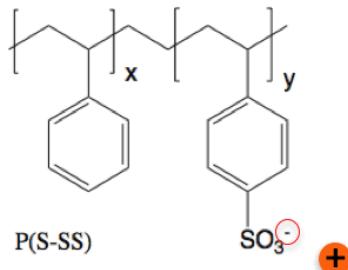
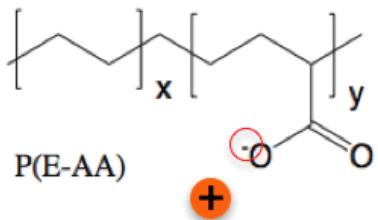
Nafion™



K. D. Kreuer, 2003

Ionomers

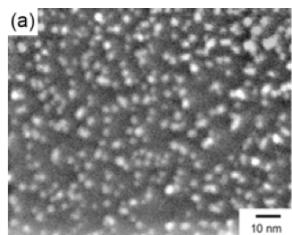
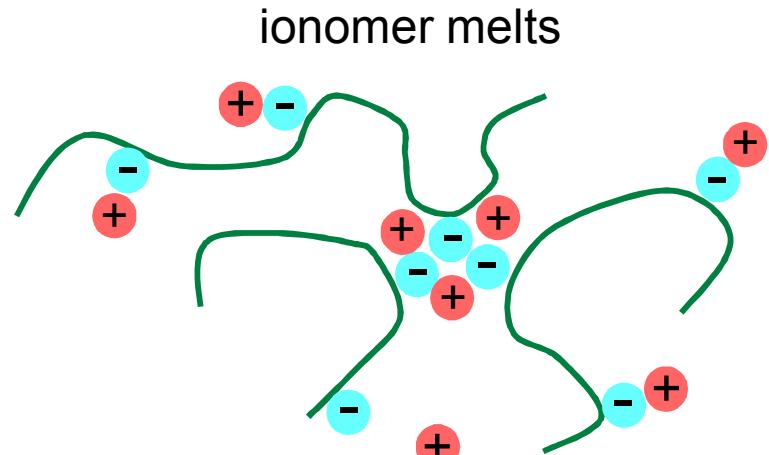
polymers with covalently-bonded ionic groups



two main classes of ionomers

- dry (melt): no solvent
- hydrated: with water

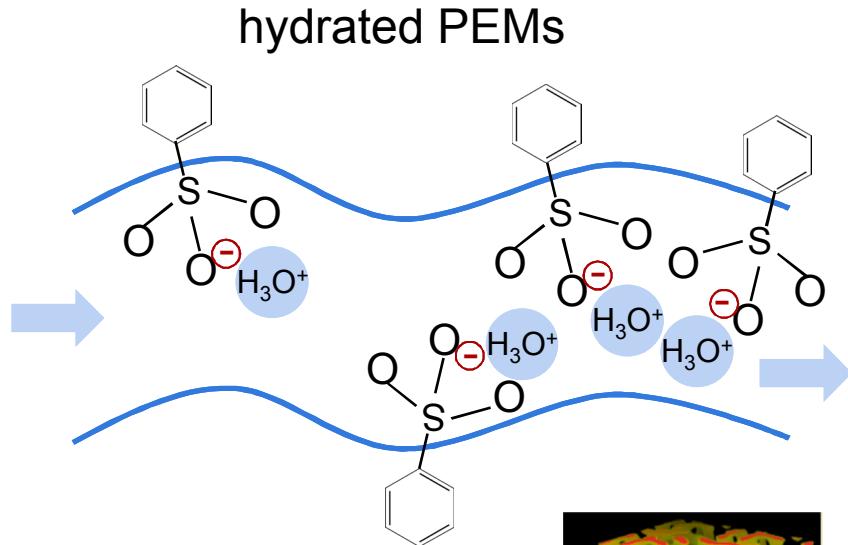
Nanoscale phase segregation



STEM showing
aggregates

PEpAA_{9.5}-Zn56

Seitz et al., *J Am Chem Soc* 132, 8165 (2010)



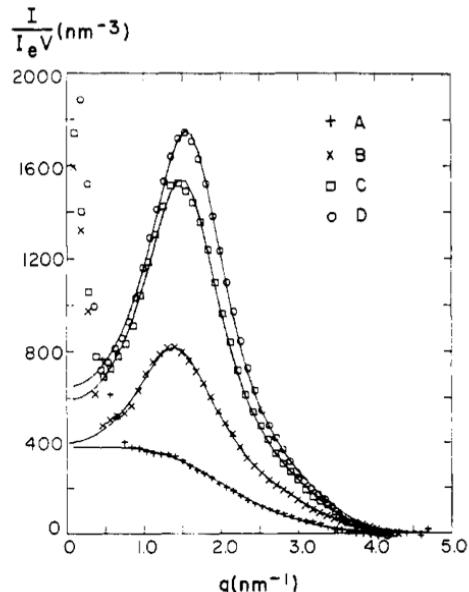
Li, N. & Guiver, M. D. *Macromolecules* 47, 2175 (2014)

Ionic aggregates in ionic polymers

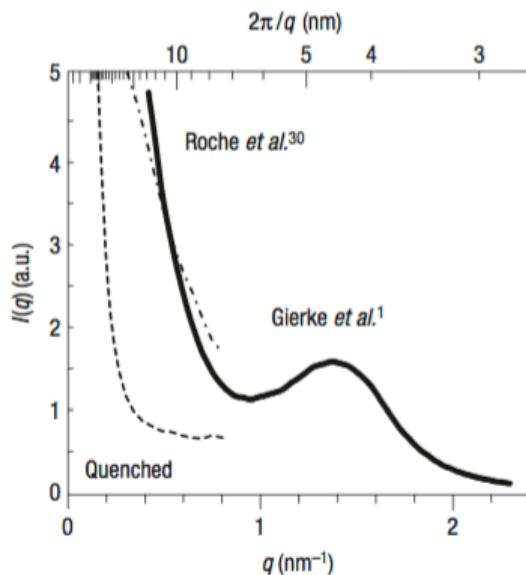
“ionomer peak”

- ubiquitous
- low wavevector peak in scattering
- from inter-aggregate scattering

PSS with Zn^{+2}



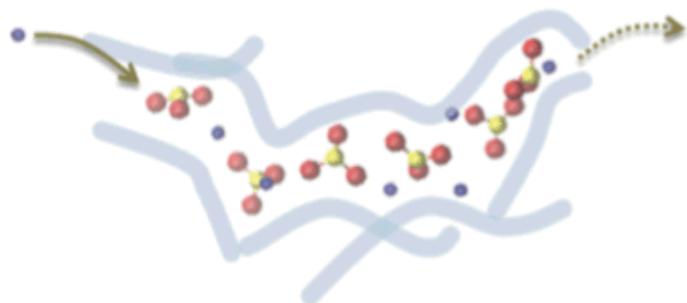
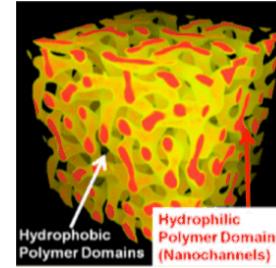
Nafion in water



How does morphology affect ion transport?

Nafion et al:
need percolated channels for H⁺ transport

“dry” ionomers
connected channels also useful
superionic transport?

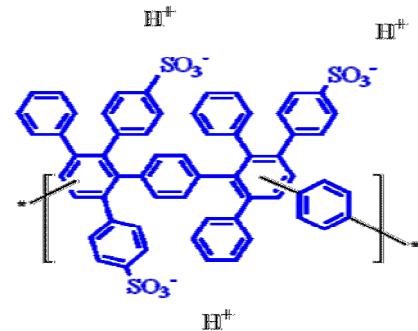
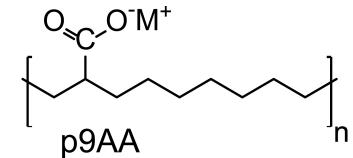


- what is the morphology?
 - simulations can help
- how do ions move?
 - decoupled from polymer?

Rest of the talk

Two ionomer systems:

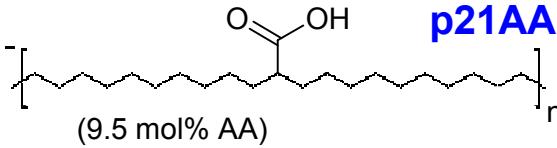
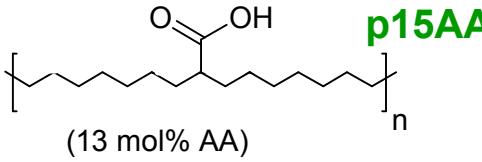
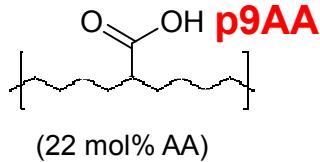
- melts of precise poly(ethylene-co-acrylic acid) (pAA)
 - transports Li^+
- hydrated, sulfonated Diels-Alder polyphenylenes (SDAPP)
 - transports H^+

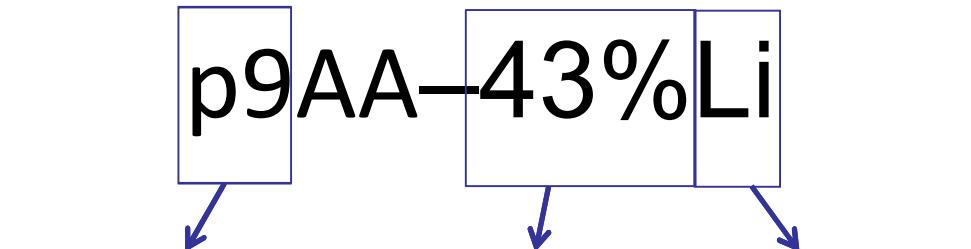


Theme: morphology of ionic aggregates/clusters

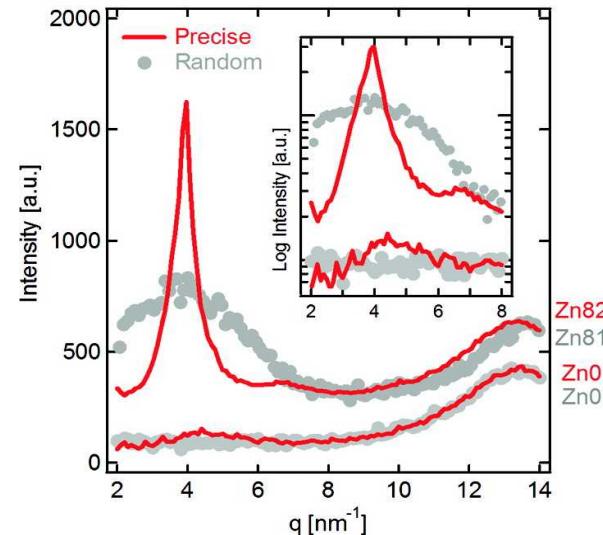
Model Materials: Precise Ionomers

PE backbone with **precisely** spaced carboxylic acid functional groups



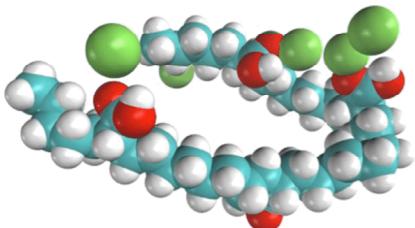


p9AA — Precise spacer length (p9, p15, p21)
43% Li — Neutralization level
Counterion type (Li⁺, Na⁺, Cs⁺, Zn²⁺)

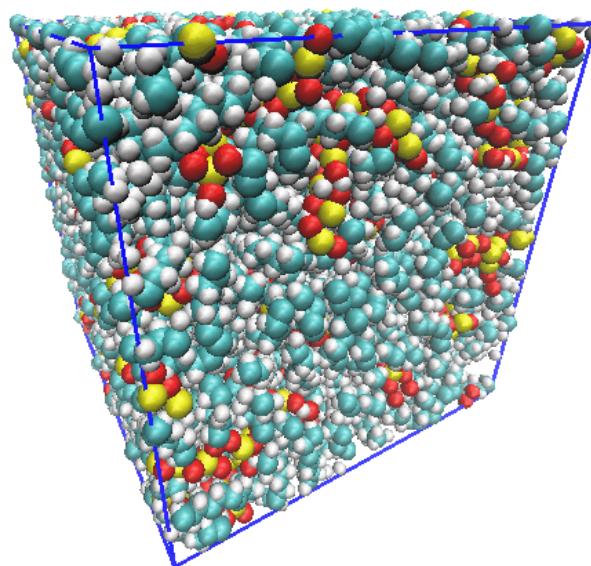


Atomistic MD simulations

- All atom L-OPLS force-field
- 80-200 polymers, 81, 90, or 84 backbone carbons/polymer
- $\sim 64 \text{ \AA}$ box, total of $\sim 25,000$ atoms
- NVT ensemble, **T well above T_g**
- LAMMPS



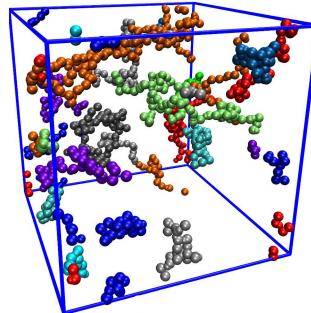
p9AA-100%Li



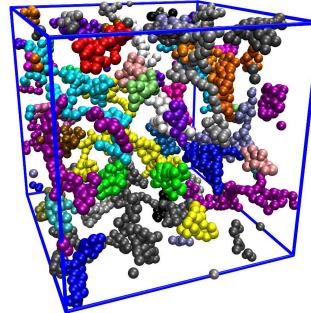
Bolintineanu et al, ACS Macro Lett, 2013

Morphology: Li-neutralized pAA

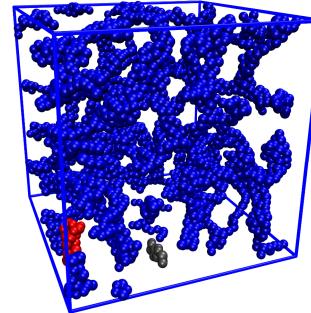
p9AA-10%Li



p9AA-43%Li

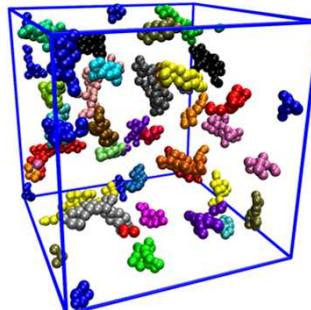


p9AA-100%Li



percolated

p21AA-43%Li

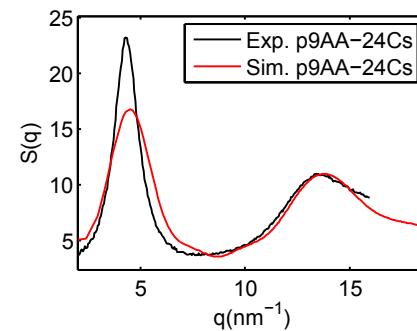
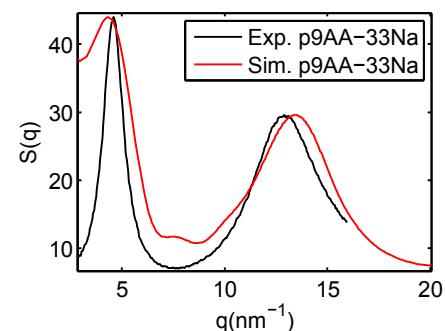
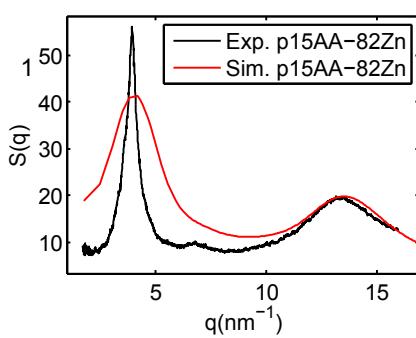
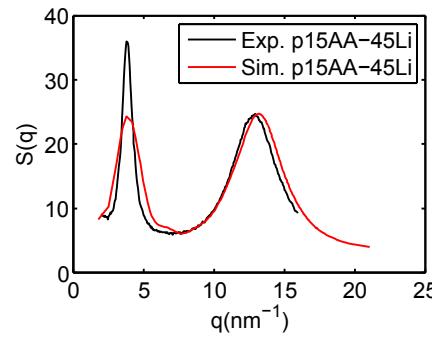
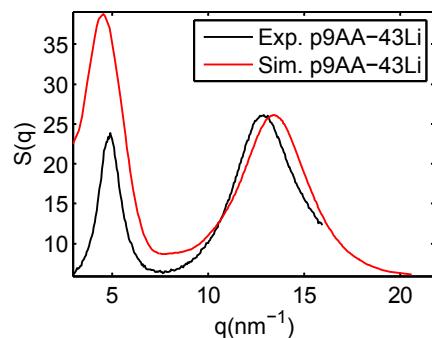


coloring by cluster

Bolintineanu et al, *ACS Macro Lett*, 2013

Direct Comparison to X-ray Scattering

$T = 393K$

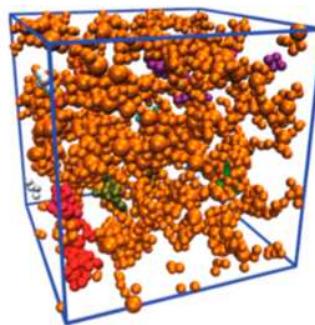
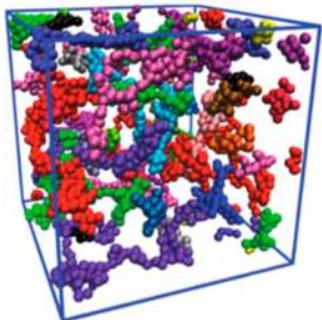
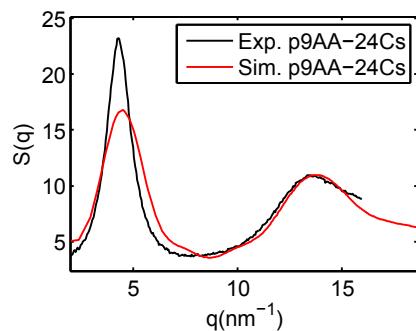
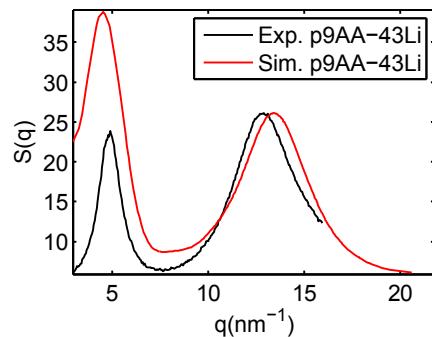


- excellent agreement in peak positions
- good agreement in peak shapes

Buitrago et al, MM 48,1210 (2015)
Seitz et al, JACS 132, 8165 (2010)

Direct Comparison to X-ray Scattering

T = 393K



- can't tell morphology from scattering
- MD simulations part of characterization

Buitrago et al, MM 48,1210 (2015)
Seitz et al, JACS 132, 8165 (2010)

Dynamics: comparison with QENS

quasi-elastic neutron scattering

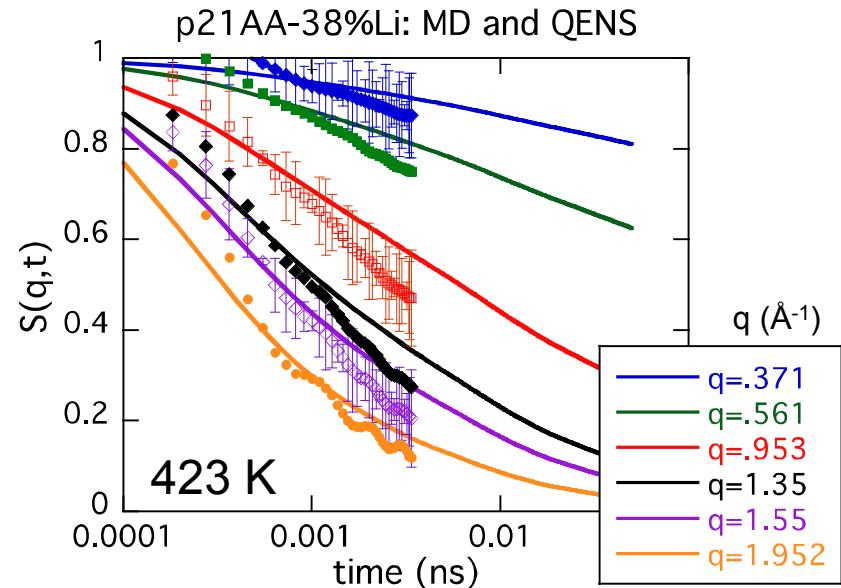
QENS limited to short times

from MD:

$$S(Q, t) = \int G_s(r, t) \frac{r \sin(Qr)}{Q} dr$$

from QENS:

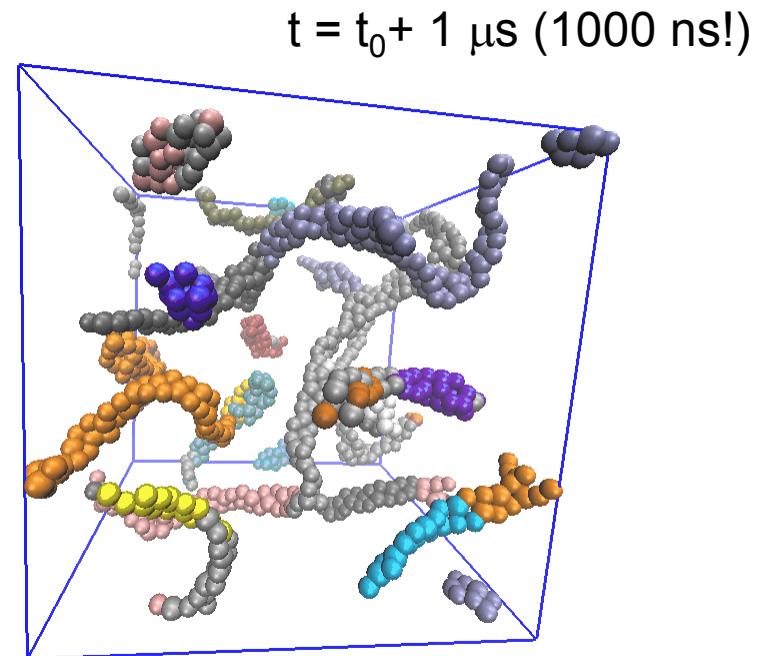
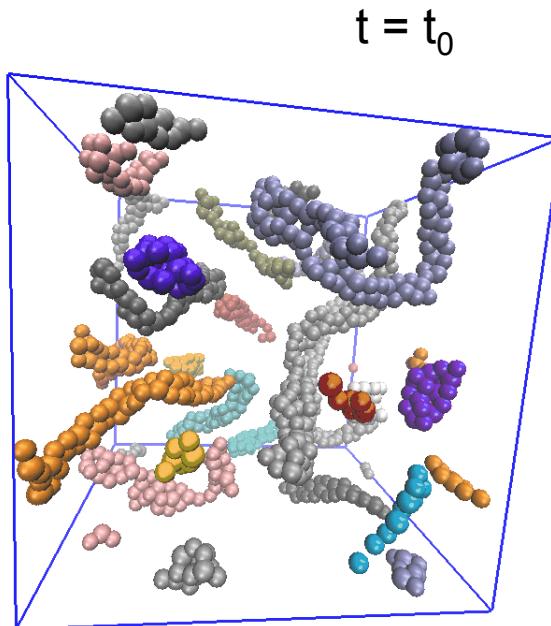
$$S(Q, t) = \frac{\int S_{exp}(Q, \omega) e^{i\omega t} d\omega}{\int R(Q, \omega) e^{i\omega t} d\omega}$$



reasonable agreement at 423K
slow: do simulations at 600K

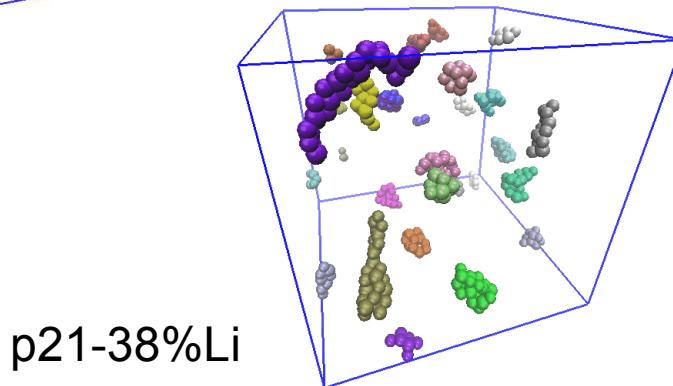
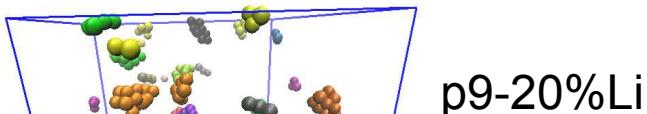
Cluster rearrangement in long simulations

p21-100%Li, T = 600K
clusters showing only Li⁺ and O⁻

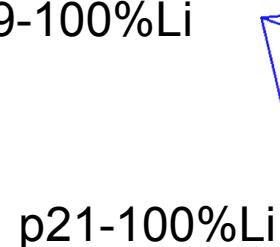
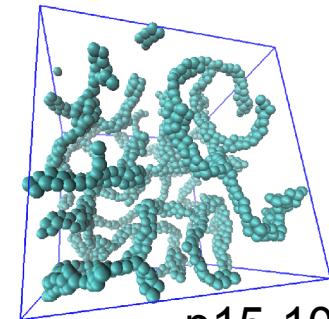
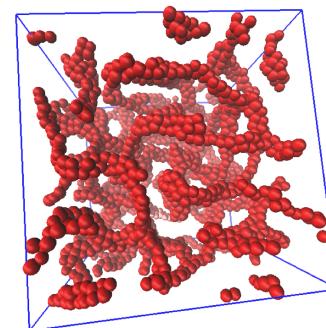


Precise ionomer morphologies

partially neutralized not percolated

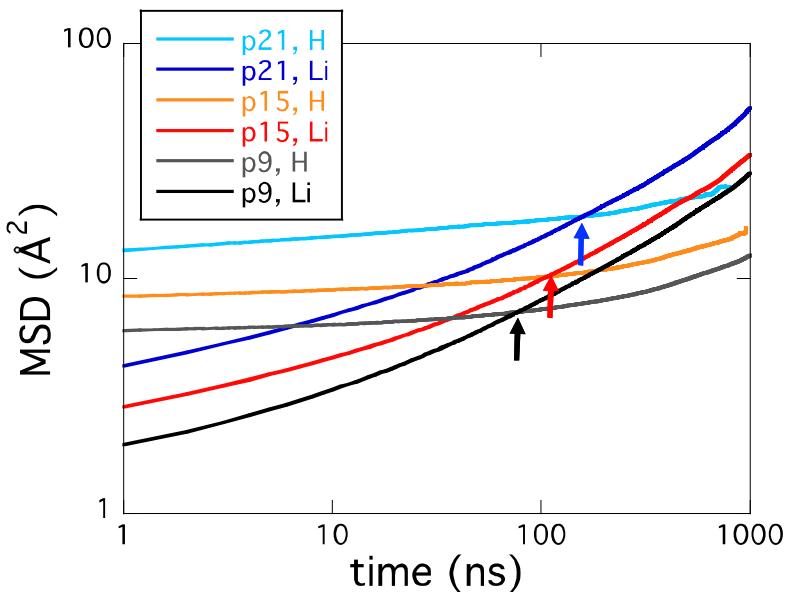


100% neutralized systems form percolated aggregates at long times

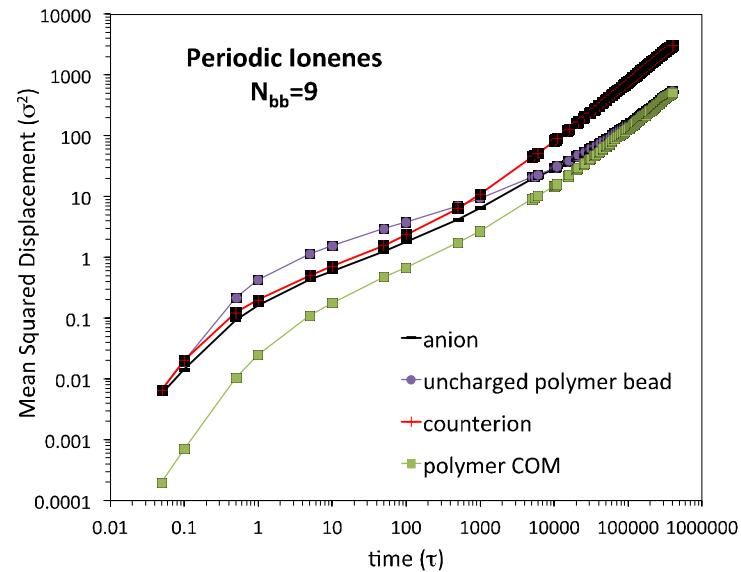


Long time dynamics

100% neutralized, H and Li MSDs

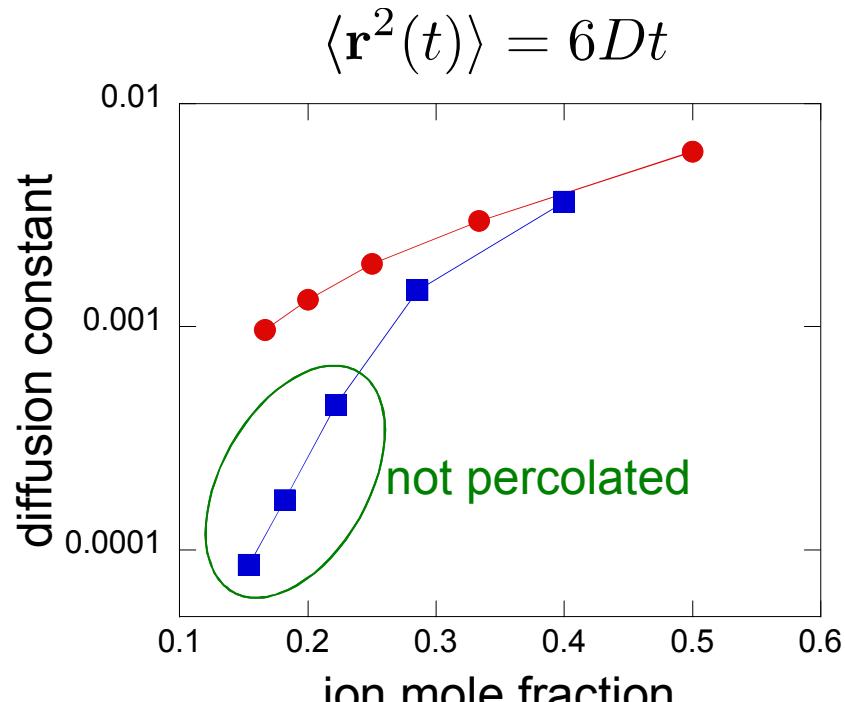
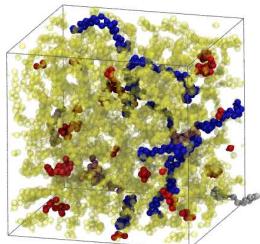
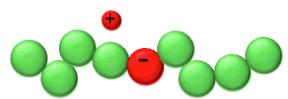
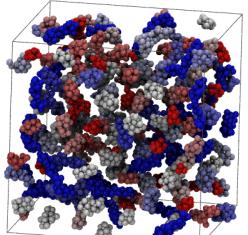
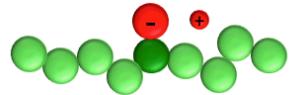


similar to CG model

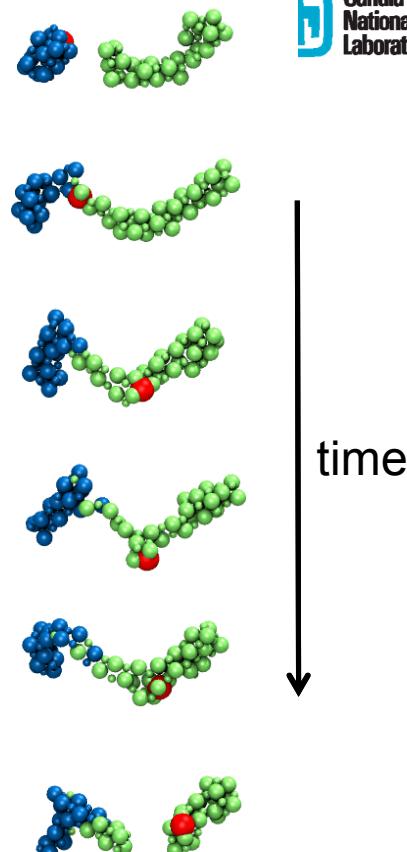


Li^+ MSD crosses H MSD first in highest ion content system

Ion motion in CG model

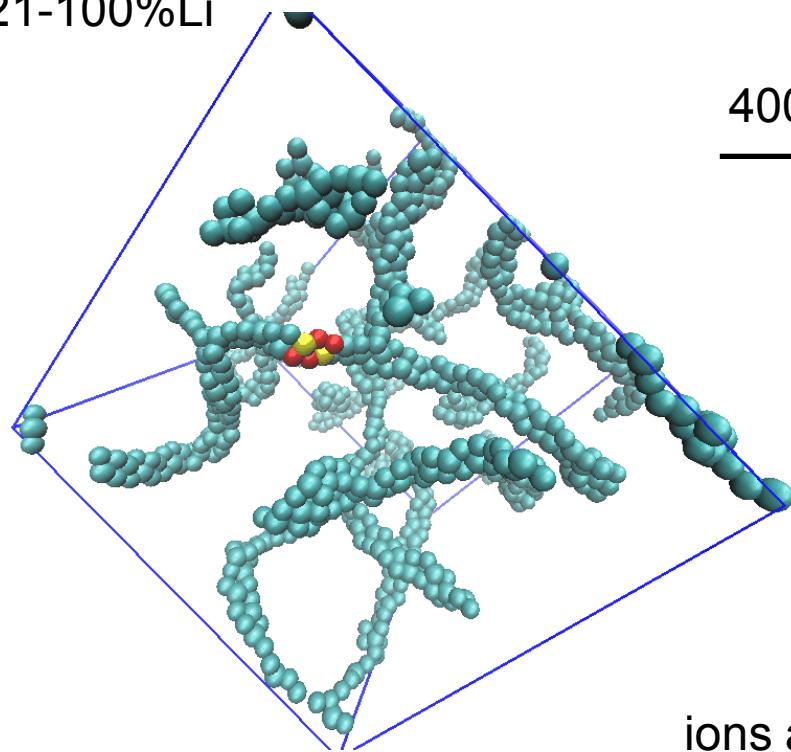


Percolated systems have faster diffusion

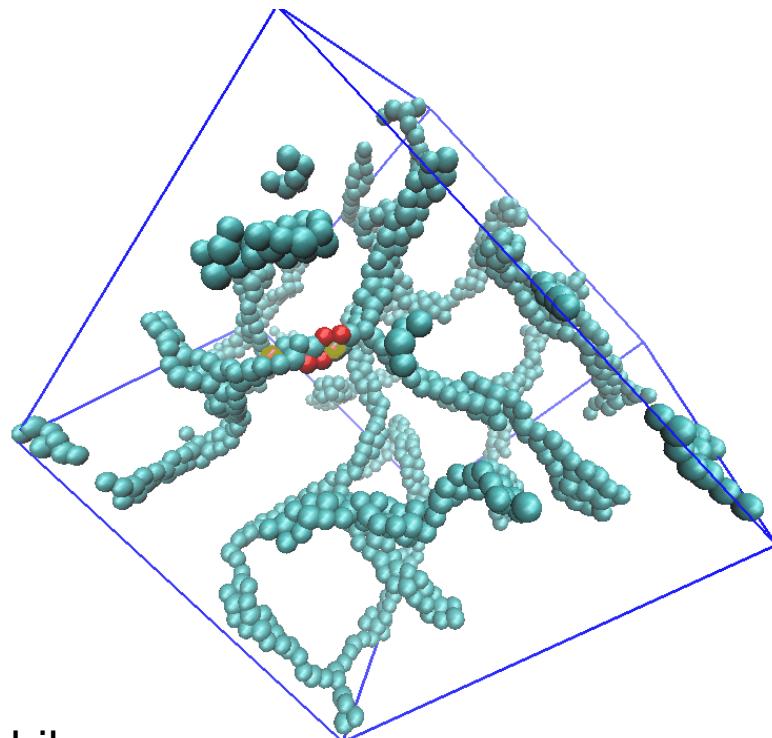


Li ions diffuse along aggregates

p21-100%Li



400 ns

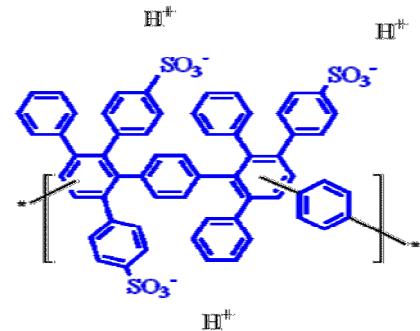
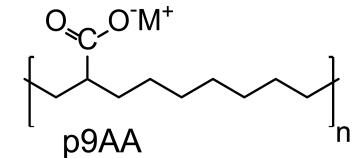


ions are mobile
very slow but similar to CG models

Rest of the talk

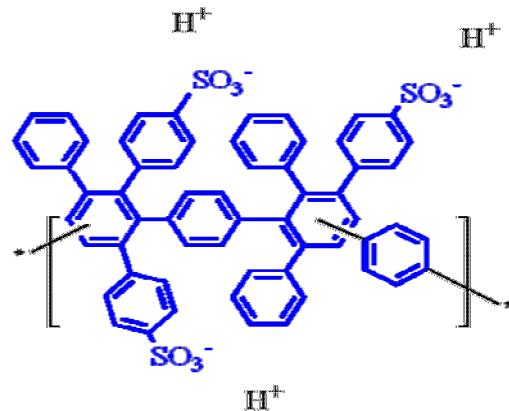
Two ionomer systems:

- melts of precise poly(ethylene-co-acrylic acid) (pAA)
 - transports Li^+
- hydrated, sulfonated Diels-Alder polyphenylenes (SDAPP)
 - transports H^+

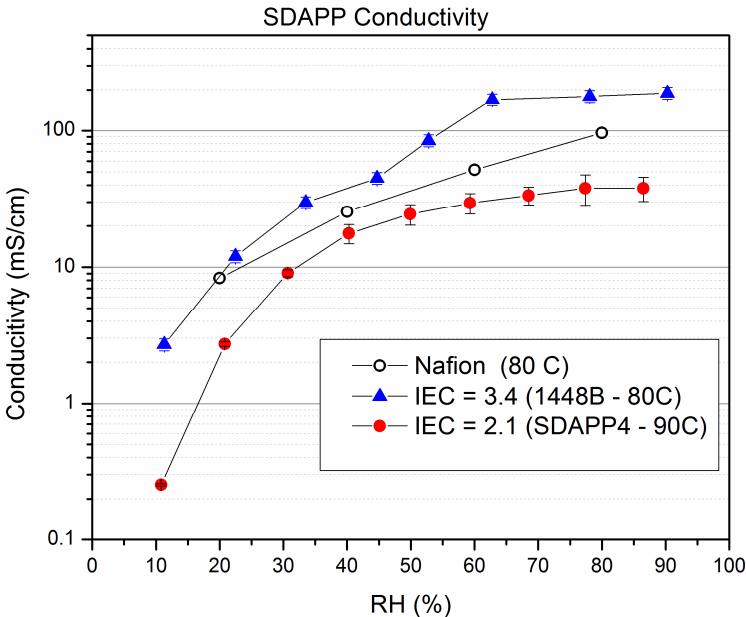


Theme: morphology of ionic aggregates/clusters

SDAPP Membranes



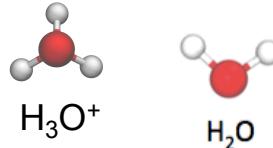
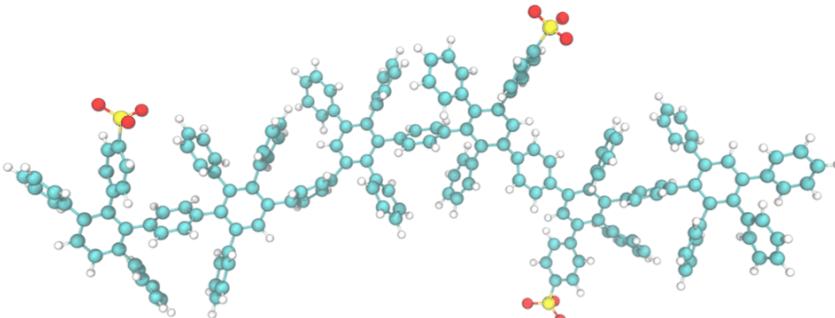
- high T_g
- high modulus
- high thermomechanical stability
- high conductivity



what is the morphology?
can we understand conductivity?

SDAPP simulations

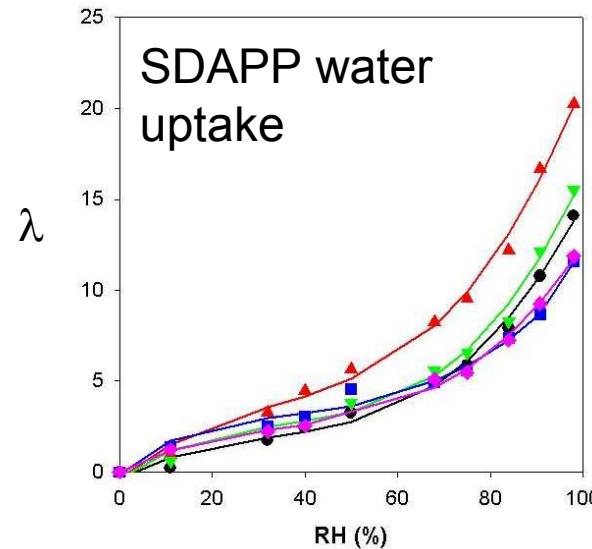
short SDAPP chain



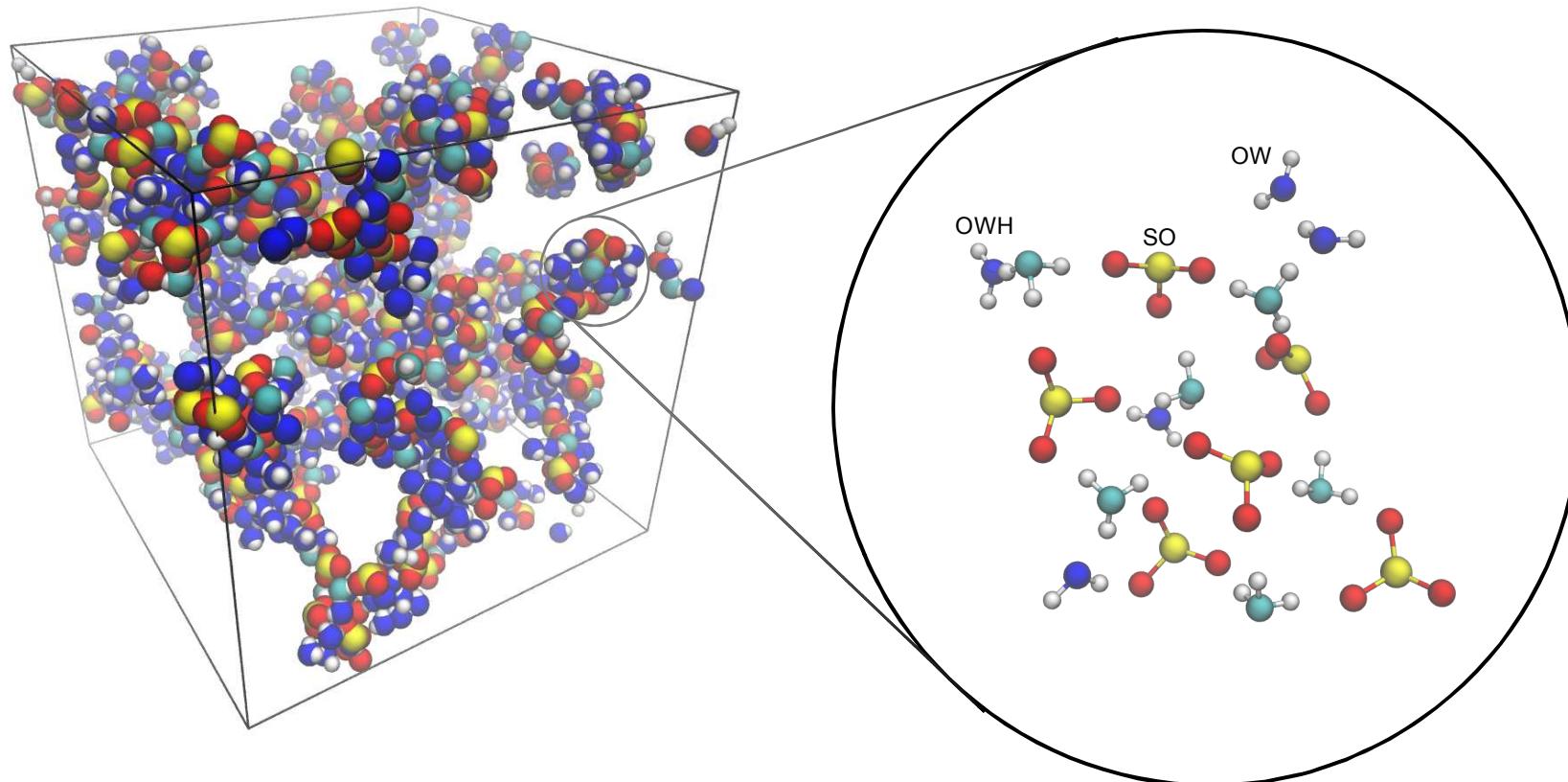
yellow = sulfur
red = oxygen
cyan = carbon
white = hydrogen

70 chains
3 monomers/chain
box size about 60Å

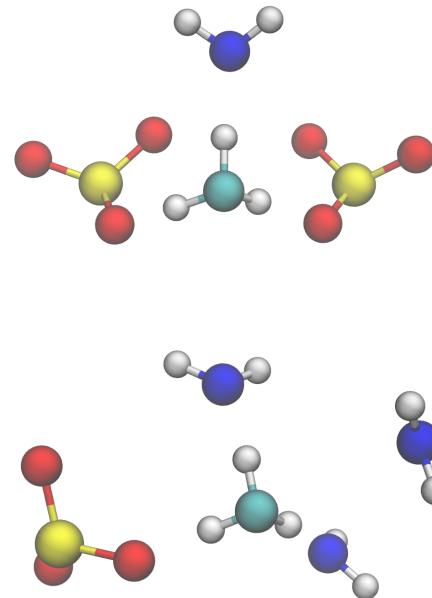
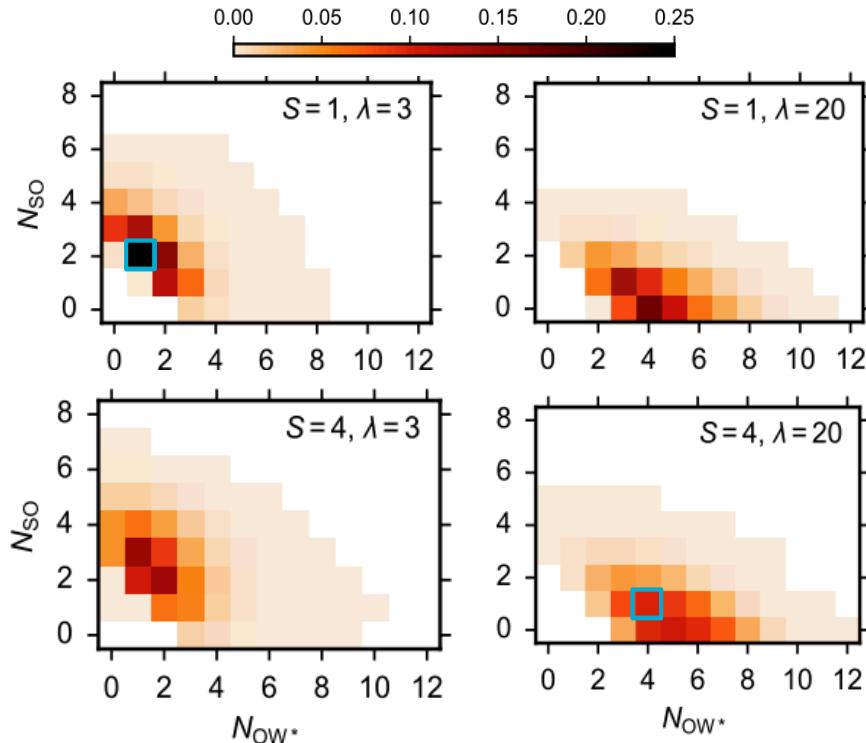
sulfonic acids/monomer:
 $S = 1, 2, 4$
waters/sulfonic acid
 $\lambda = 3, 5, 10, 20$



Ion/water aggregates formed



Hydronium ion coordination



$N_{OW^*} = 1, N_{SO} = 2$
(25%)

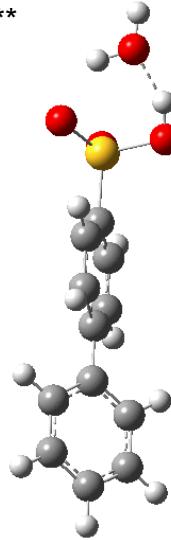
$N_{OW^*} = 3, N_{SO} = 1$
(11%)

N_{SO} = number of SO_3 in first solvation shell of H_3O

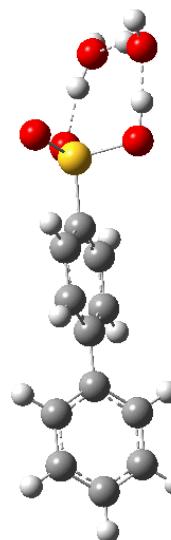
N_{OW^*} = number of H_2O, H_3O in first solvation shell of H_3O

Quantum cluster calculations

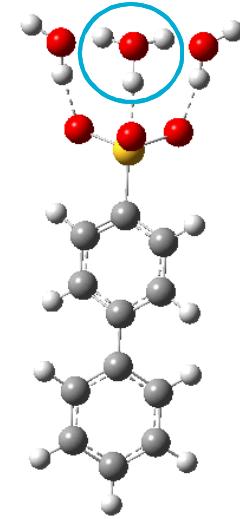
DFT 6-311**



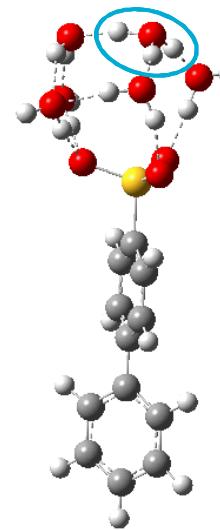
$\lambda = 1$
Hydrogen
bond



$\lambda = 2$
Hydrogen
bond



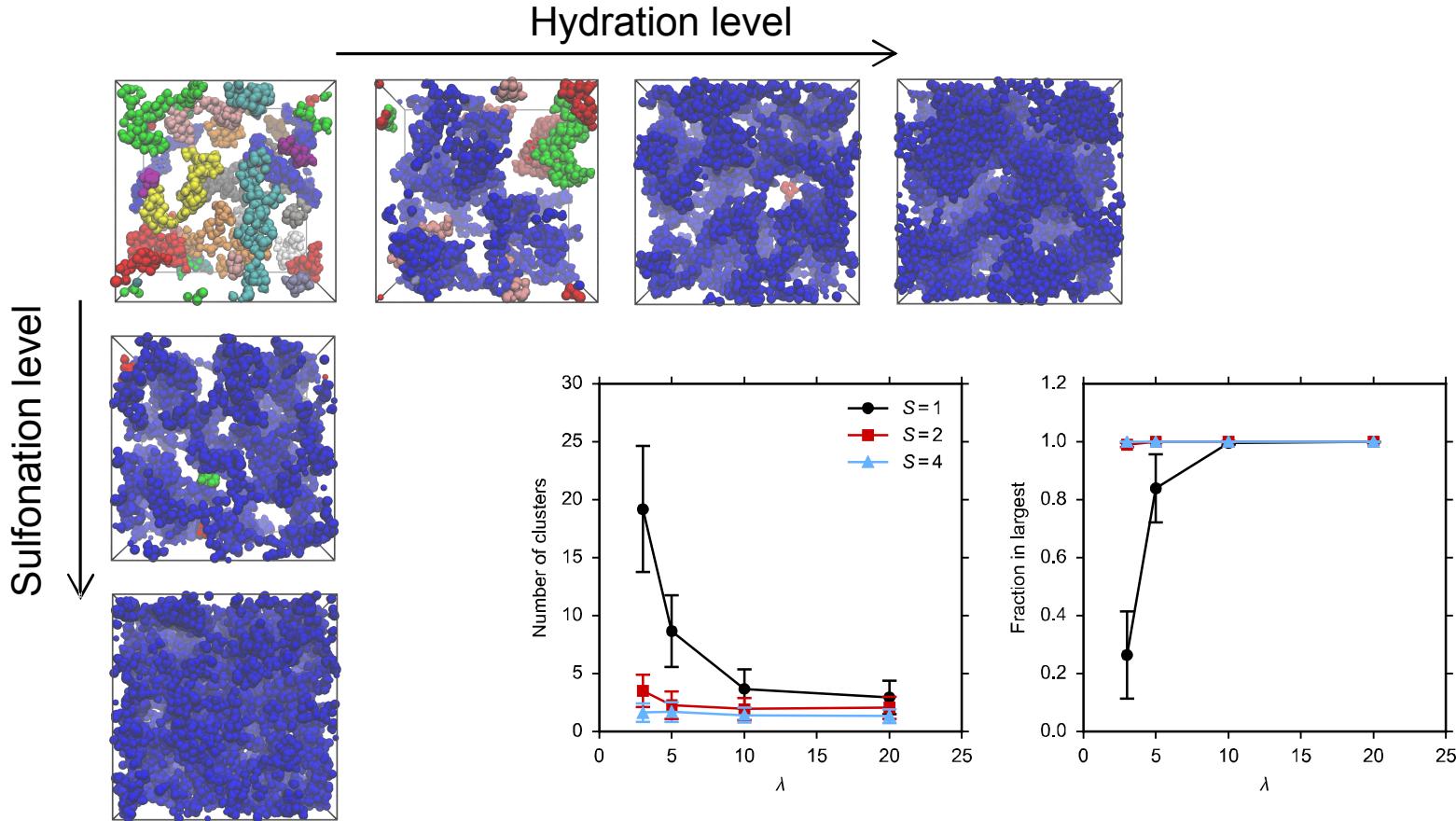
$\lambda = 3$
Contact ion
pair



$\lambda = 5$
Solvated contact
ion pair

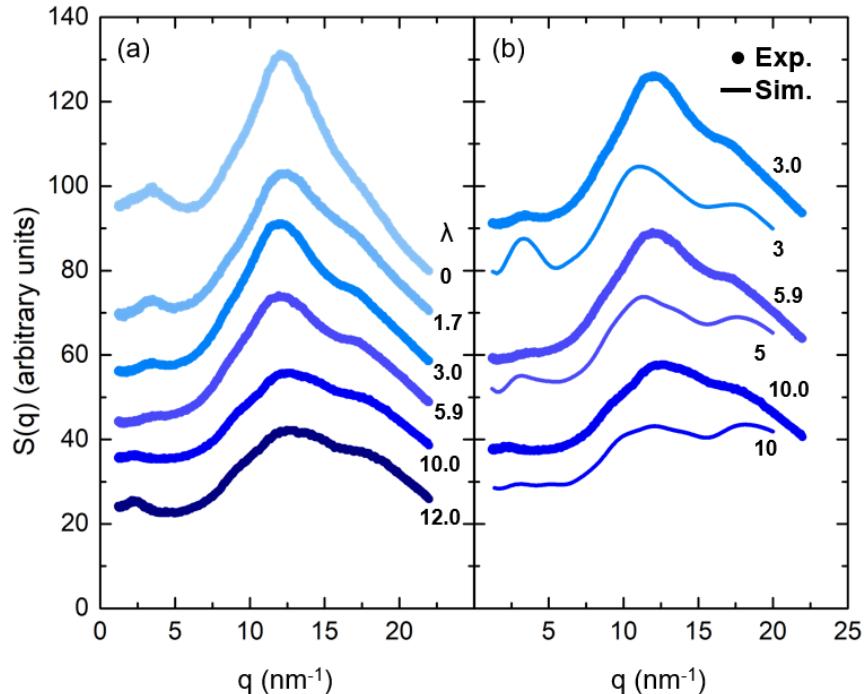
need $\lambda = 3$ for deprotonation

Distance-based cluster morphologies



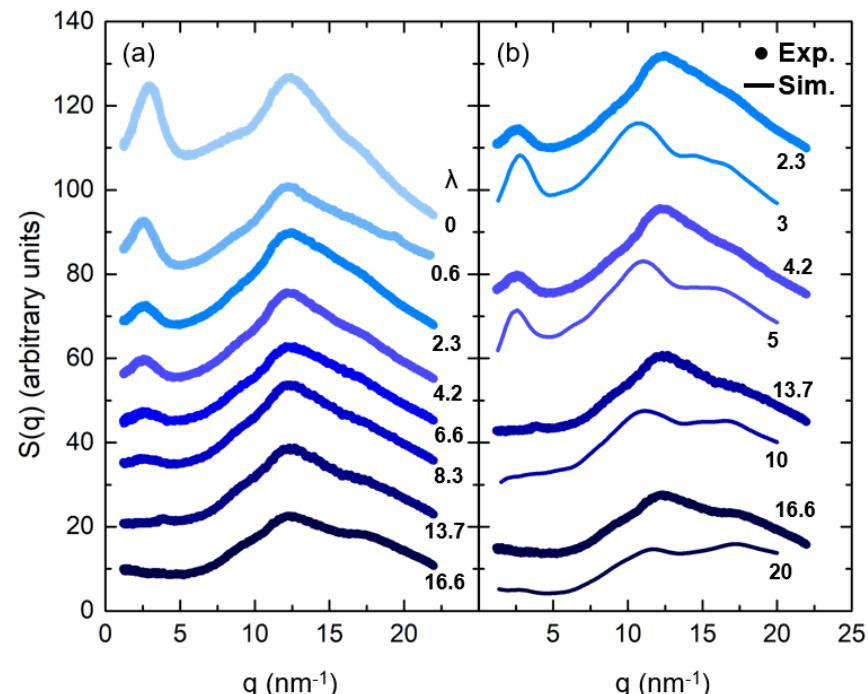
Structure Factors

experiments S=3.6



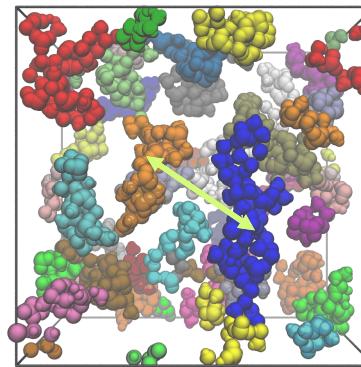
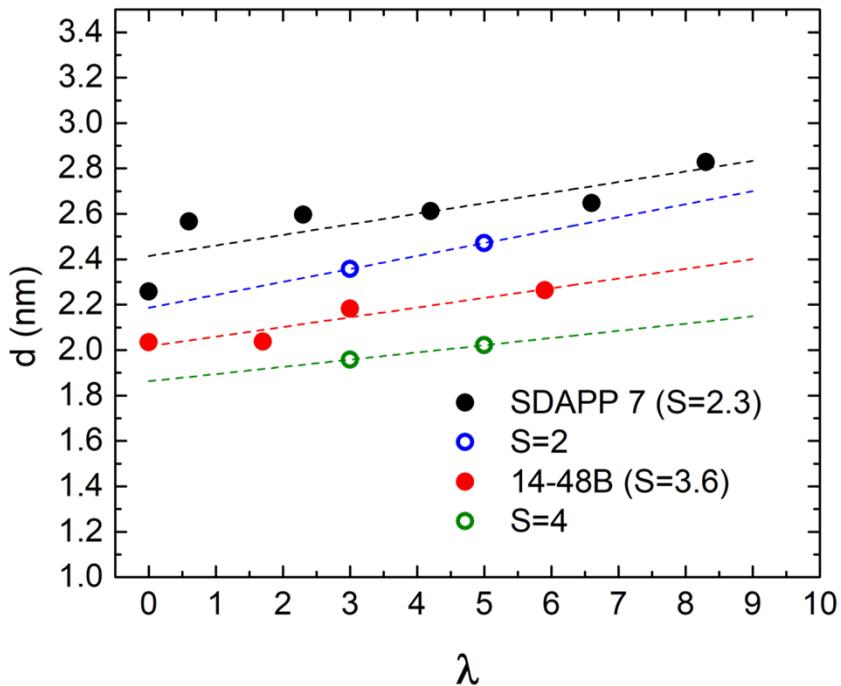
MD, S=4

experiments, S=2.3



Correlation distance between aggregates

$$d = 2\pi/q^*, q^* = \text{ionomer peak location}$$



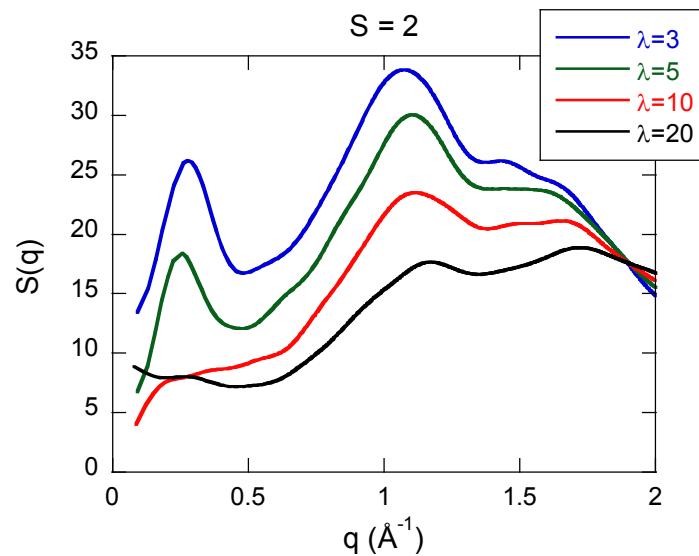
$S=1, \lambda = 3$

from real space snapshots, low λ :
 $S = 2, d = 23 \text{ \AA}$
 $S = 4, d = 19 \text{ \AA}$

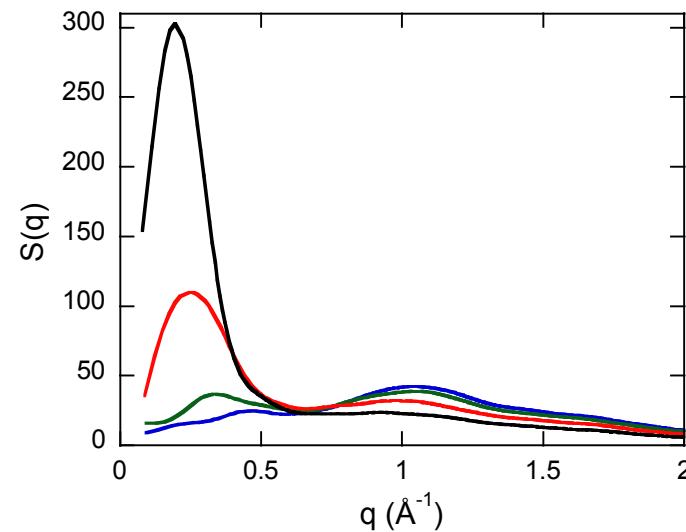
MD consistent with X-ray

Why does the ionomer peak decrease with λ ?

total $S(q)$

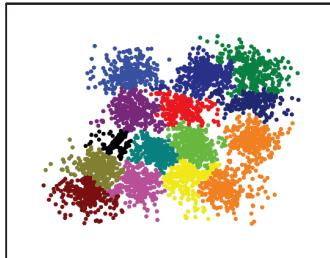
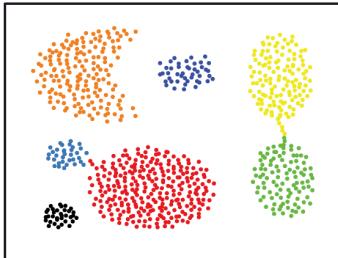


set water scattering length = 0



phase segregation in x-ray masked by loss of contrast
MD simulation helps confirm morphology!

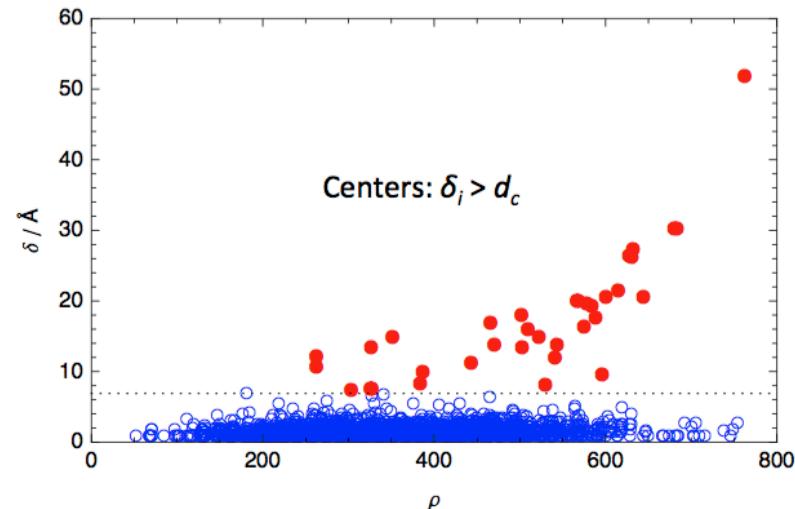
Density-based clustering



Rodríguez, A., & Laio, A. (2014). Clustering by fast search and find of density peaks. *Science*, 344(6191), 1492–1496.

Algorithm:

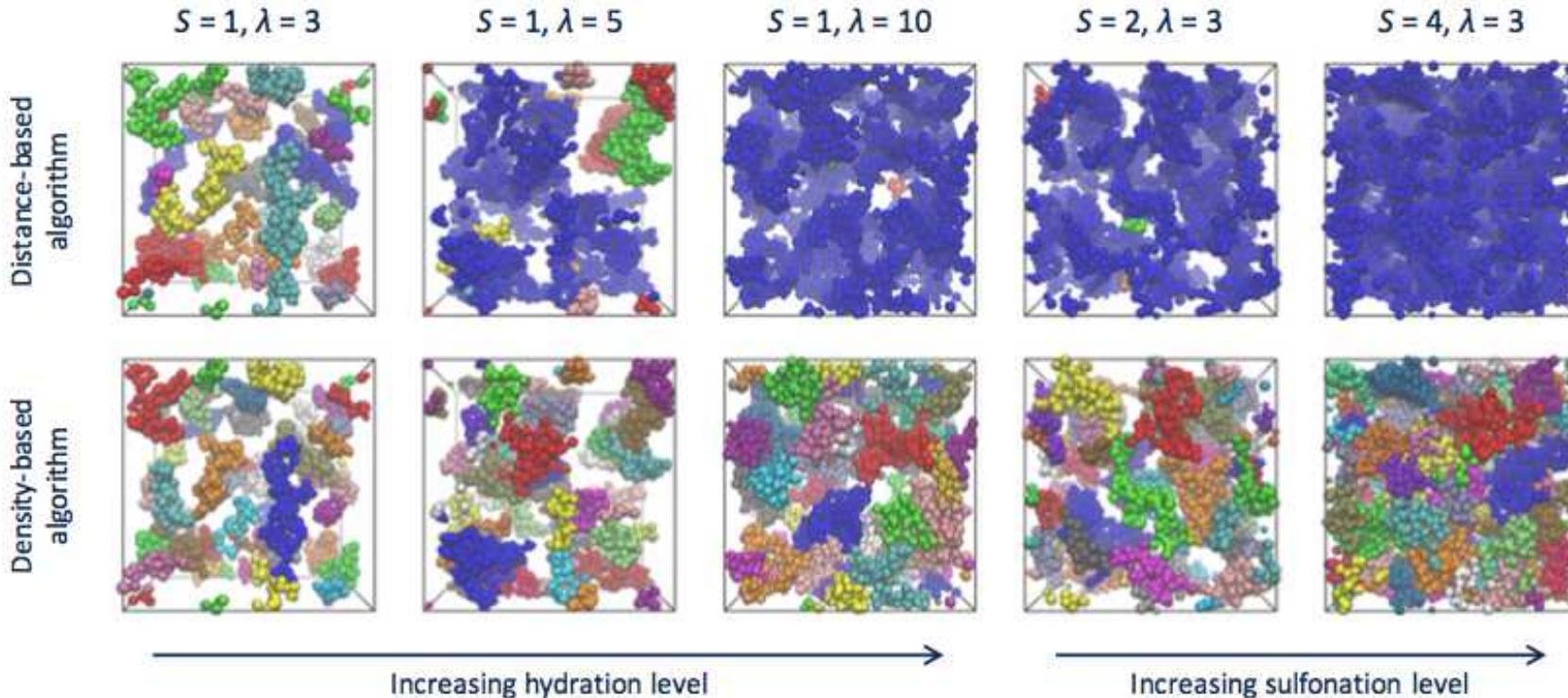
1. Each atom is assigned parameter ρ , which describes the local density as number of atoms within given cutoff distance r_c
2. Each atom is assigned parameter δ , which is the minimum distance from an atom of higher density
3. Cluster centers are chosen as atoms with highest density such that $\delta_i > \delta_{\min}$
4. Remaining atoms are assigned to same cluster as nearest neighbor of higher density



Both r_c and δ_{\min} are determined from first peak in SO-SO RDF to include first solvation shell

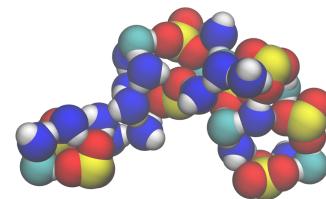
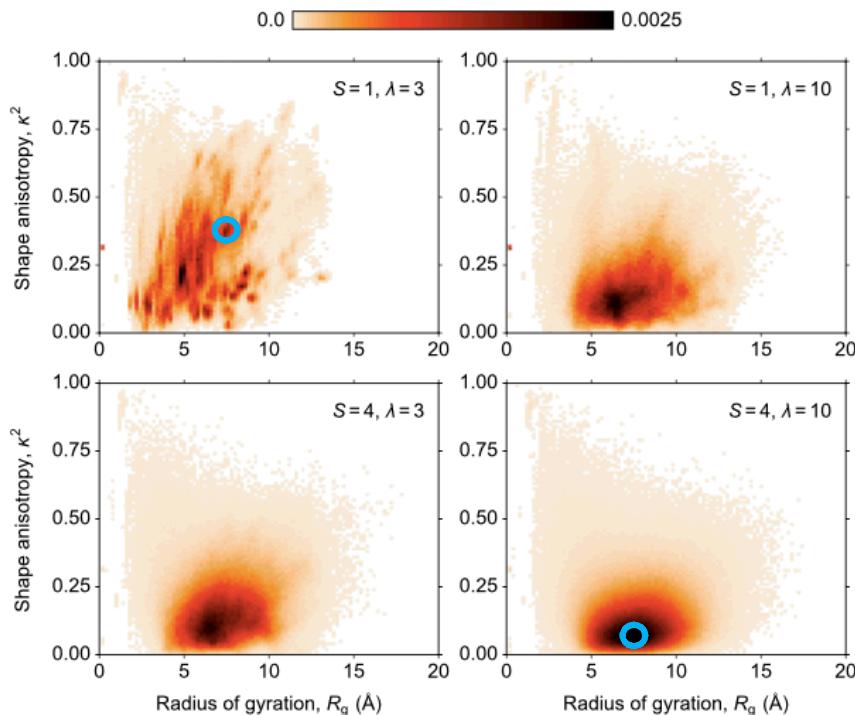
SDAPP Clusters

density-based algorithm resolves differences in percolated systems

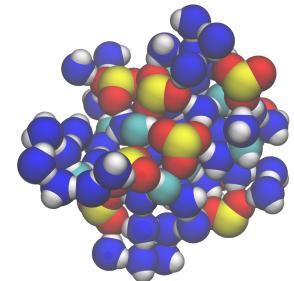


Cluster size and shape

$$R_g^2 = \lambda_1 + \lambda_2 + \lambda_3 \quad \kappa^2 = 1 - 3(\lambda_1\lambda_2 + \lambda_1\lambda_3 + \lambda_2\lambda_3)/R_g^4$$

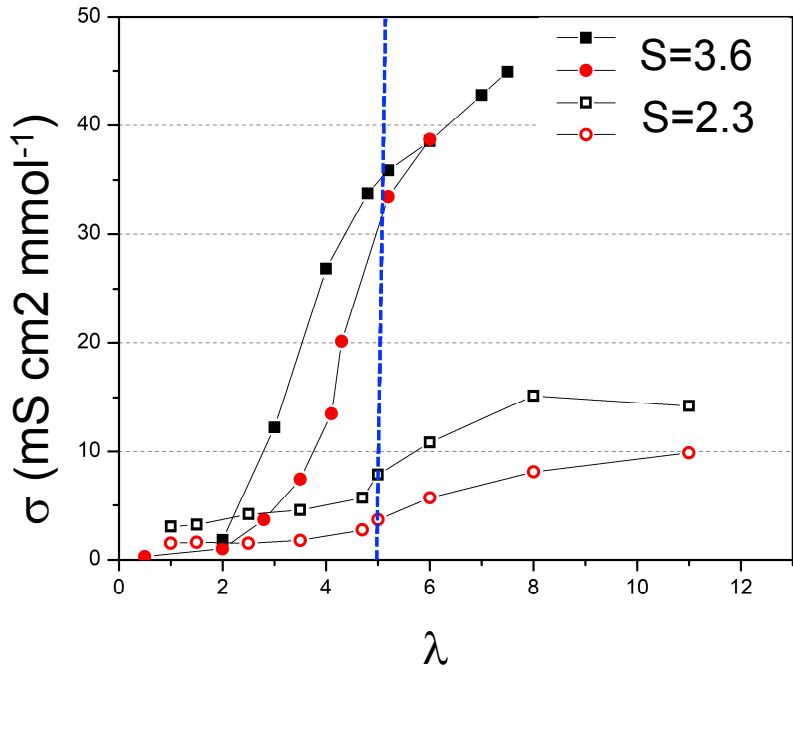


$S = 1, \lambda = 3, \kappa^2 = 0.4$

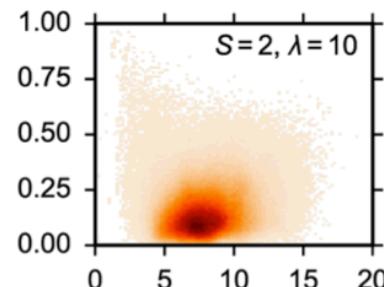
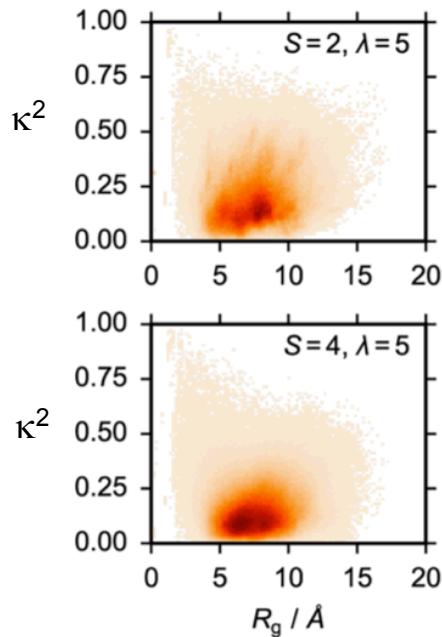


$S = 4, \lambda = 10, \kappa^2 = 0.05$

Conductivity in SDAPP



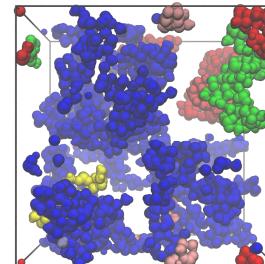
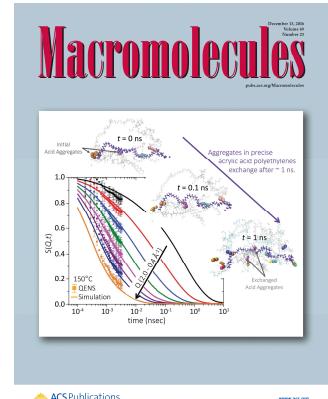
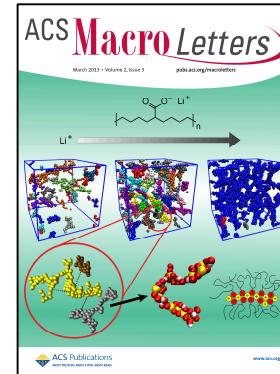
do more spherical cluster shapes improve conductivity?



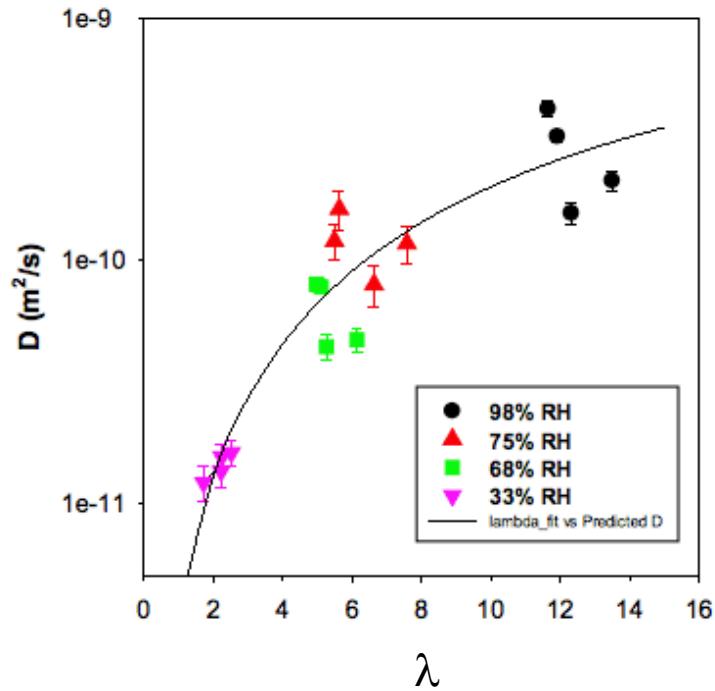
or just more waters locally in the cluster?

Summary

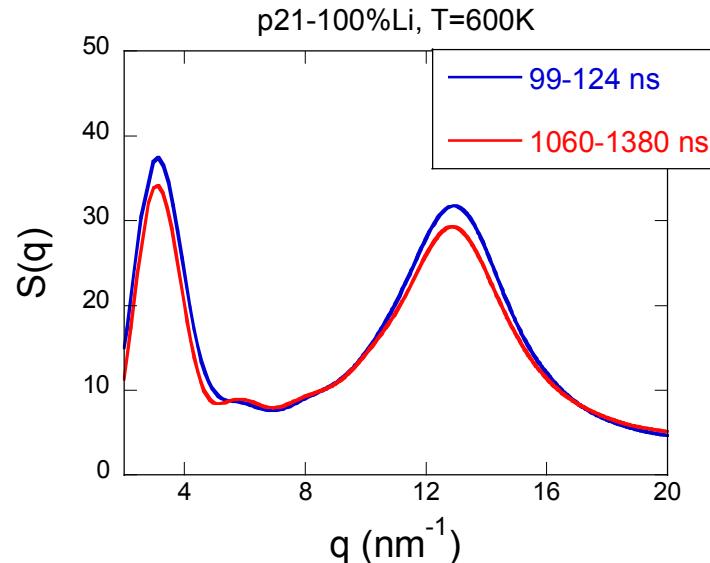
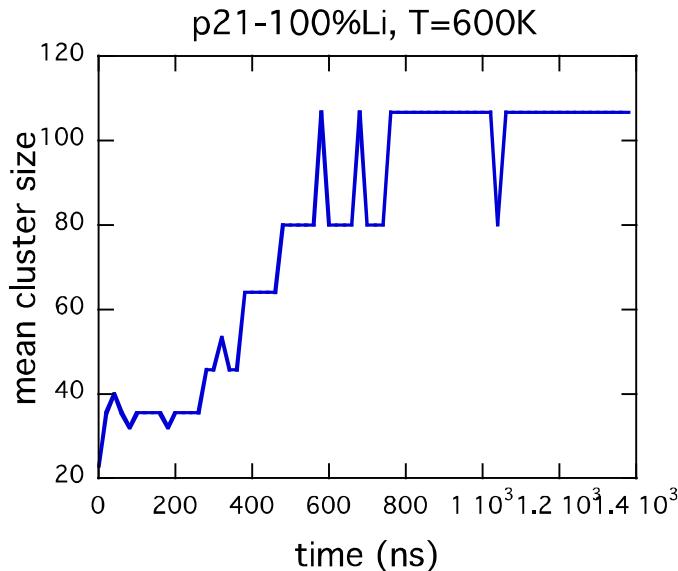
- MD for precise ionomers
 - MD agrees with X-ray, QENS
 - ions move faster in percolated aggregates
 - motion along aggregates, decoupled from polymer chains
- MD for SDAPP
 - MD agrees with X-ray
 - density-based algorithm for more information on percolated clusters
 - ion transport depends on cluster shape?



Future work: continued correlation of dynamics with morphology

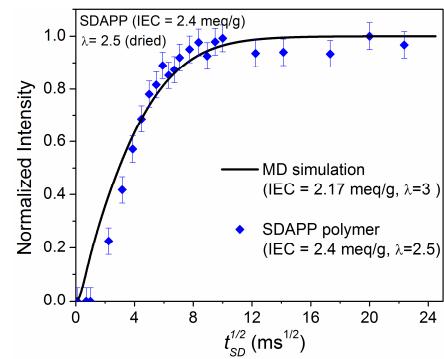
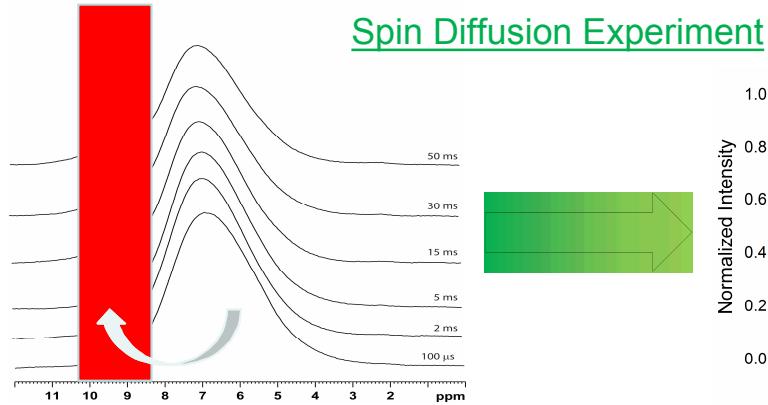
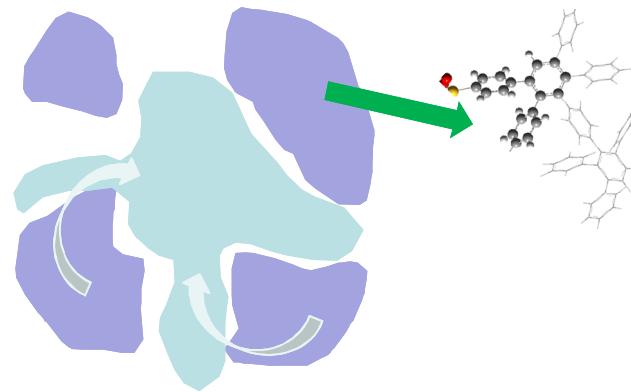
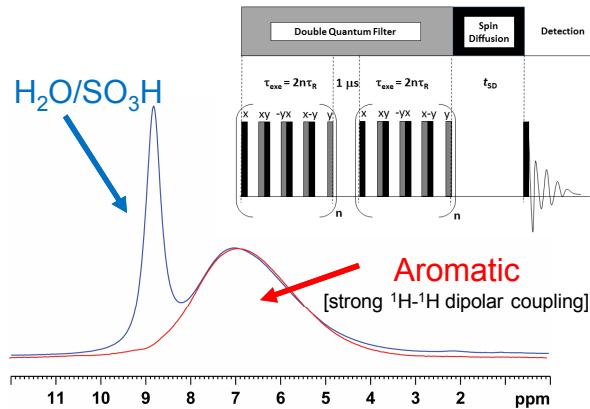


Cluster dynamics

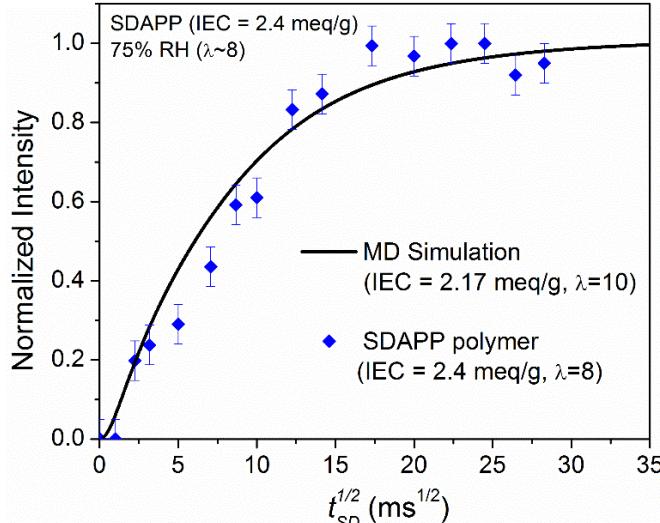
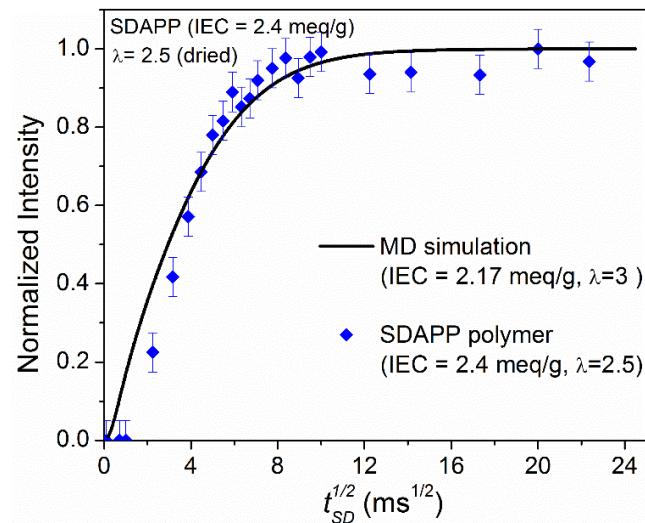
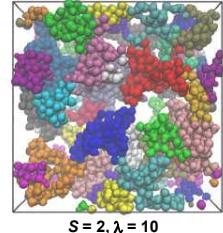
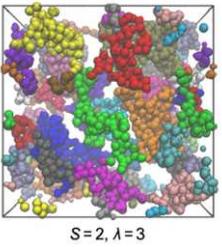


static structure not very sensitive to percolation or not

NMR Spin Diffusion



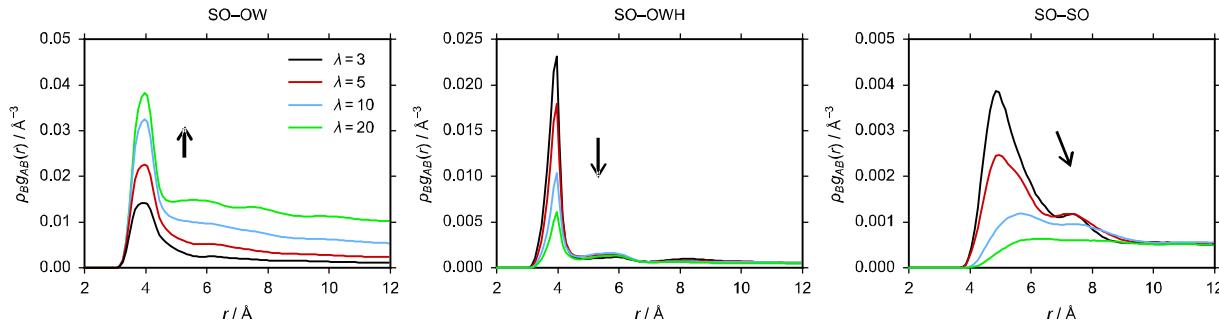
MD compared to NMR spin diffusion



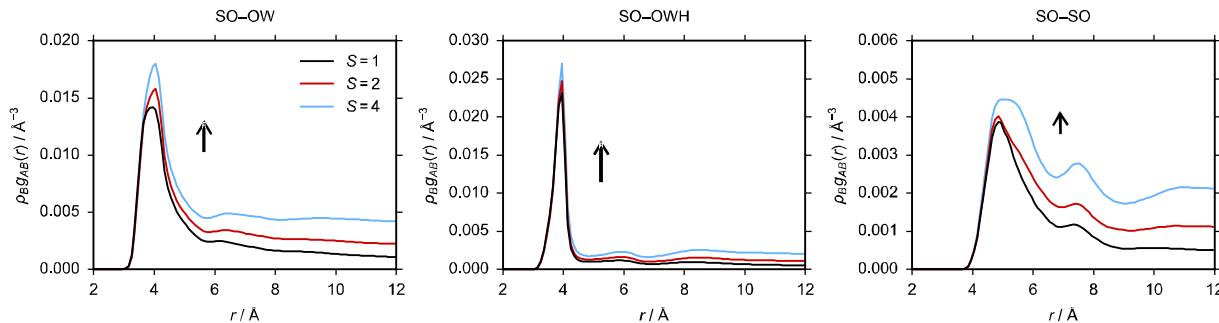
- MD structure (morphology from simulation)
- Diffusion constants, volume fractions, etc. are fixed.
- No adjustable parameters in these fits!!!!**
- Deviations at higher hydration levels (finite simulation size)

Sulfonate group more solvated at higher water contents

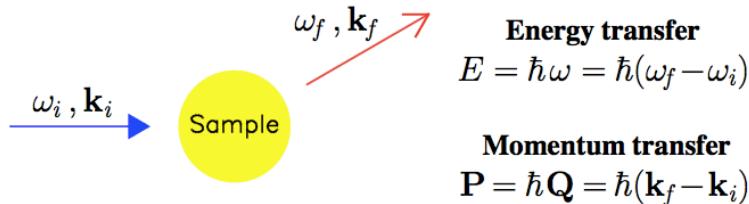
Increasing hydration level:



Increasing sulfonation level:



Quasi-Elastic Neutron Scattering

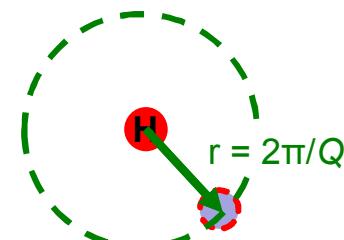


incoherent, inelastic: sensitive to self-motion of hydrogens

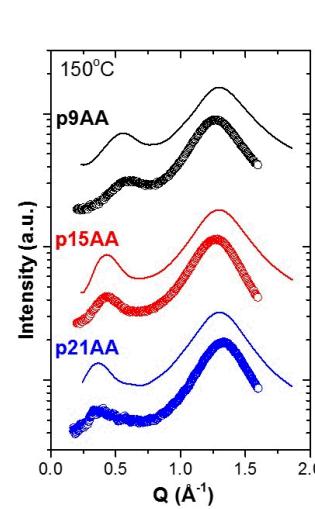
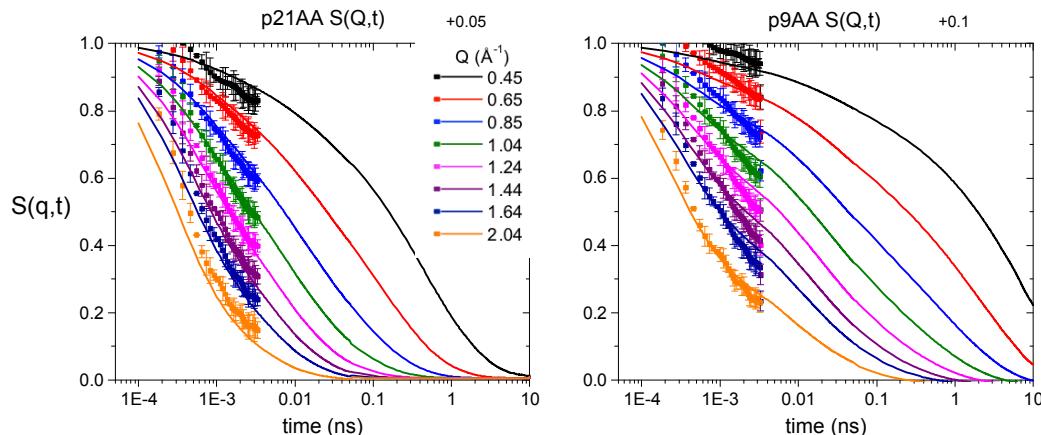
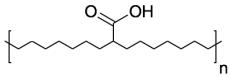
$G_s(r, t)$: given an atom was at $r=0$ at time $t=0$, the probability that the atom is at r at time t

from MD:
$$S(Q, t) = \int G_s(r, t) \frac{r \sin(Qr)}{Q} dr$$

from QENS:
$$S(Q, t) = \frac{\int S_{exp}(Q, \omega) e^{i\omega t} d\omega}{\int R(Q, \omega) e^{i\omega t} d\omega}$$



Acid Copolymers



excellent agreement between QENS and MD

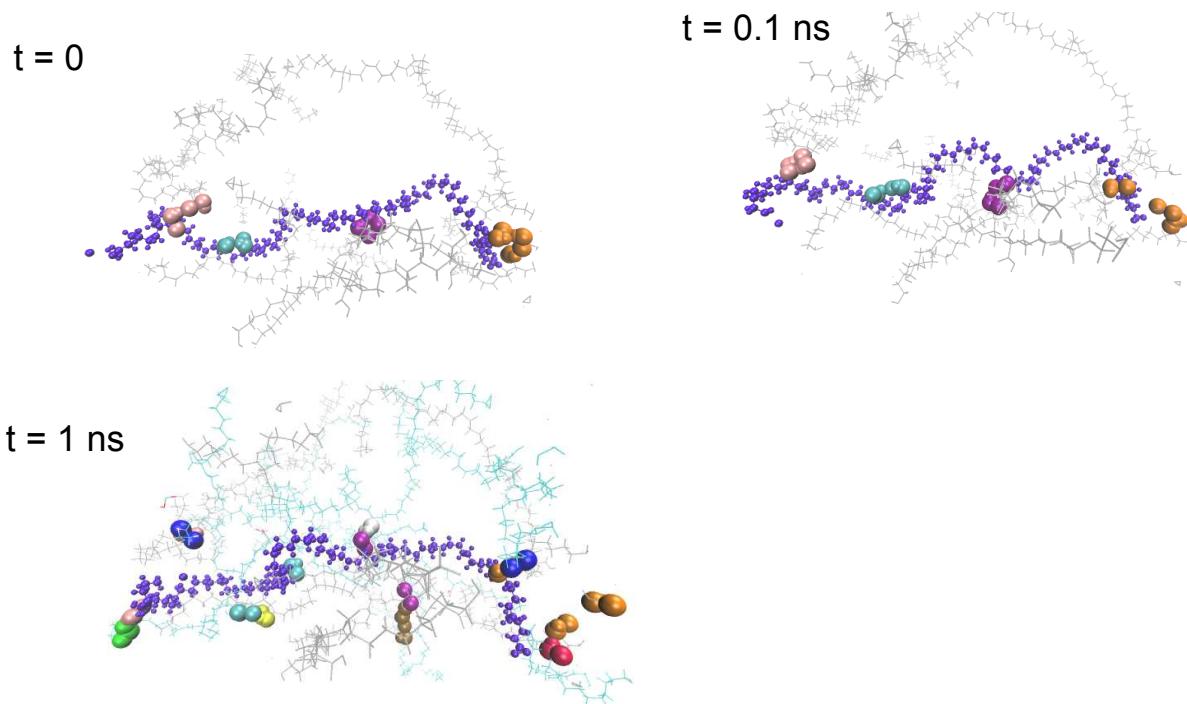
relevant length scales:

amorphous halo: $Q \approx 1.35 \text{\AA}^{-1}$

ionomer peak: $Q \approx 0.3 - 0.6 \text{\AA}^{-1}$

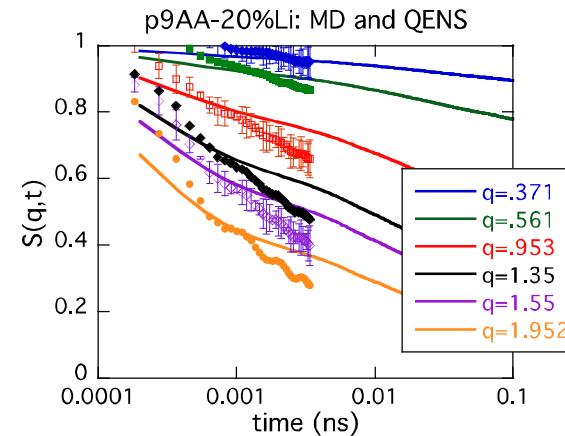
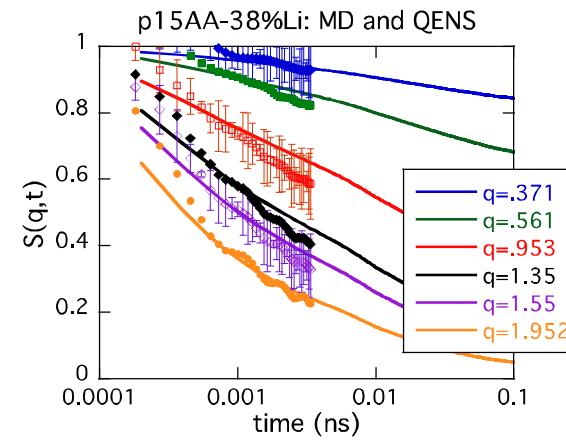
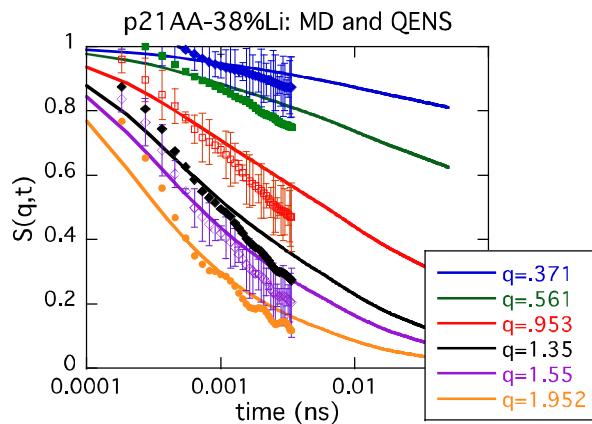
Acid aggregates rearrange

p21AA: one chain



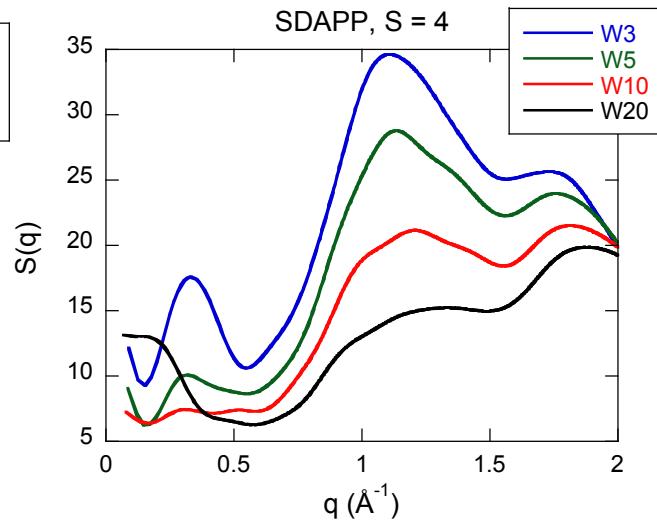
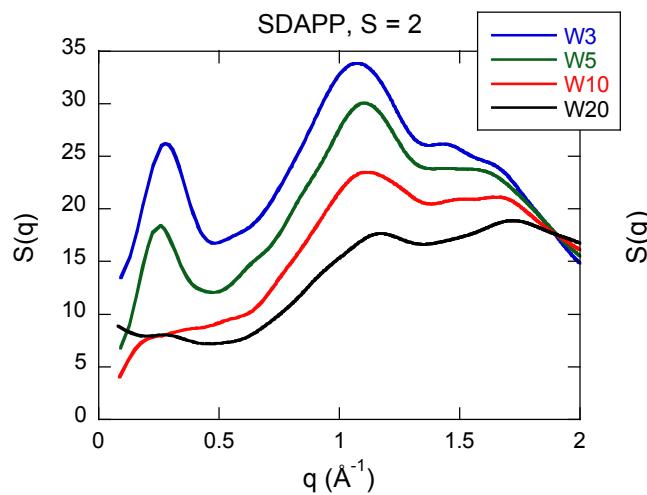
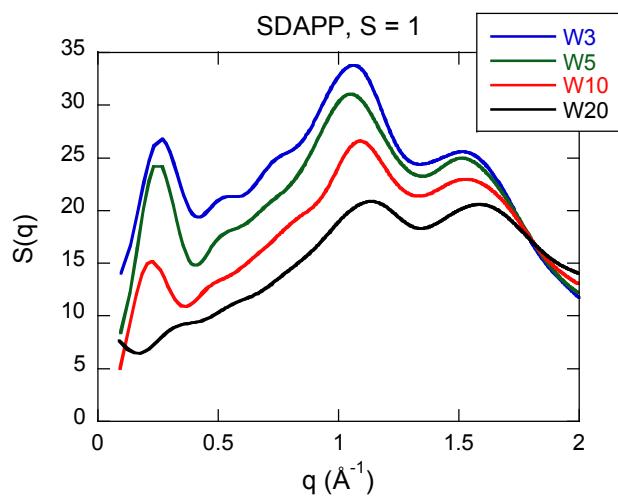
Middleton et al, *Macromolecules* **49**, 9176 (2016)

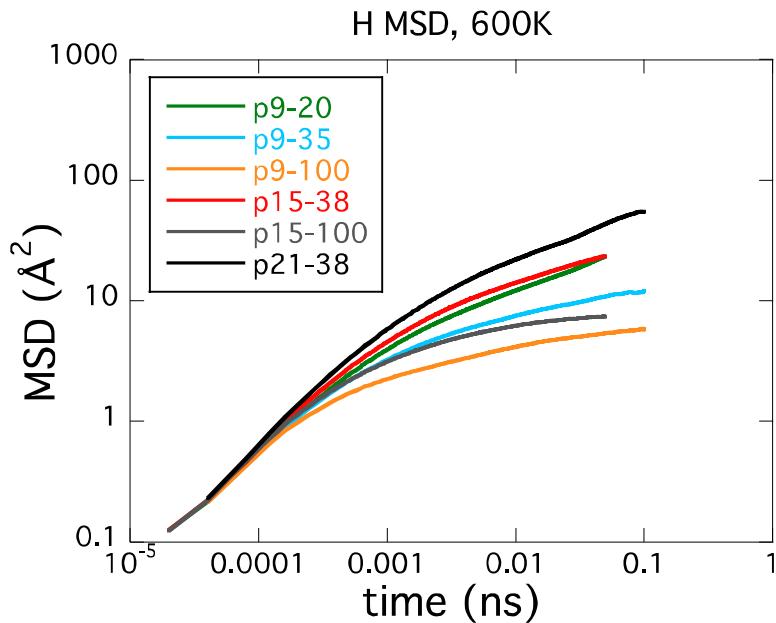
QENS vs MD



reasonable agreement

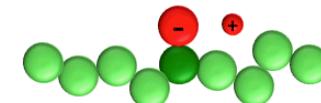
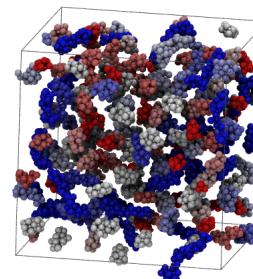
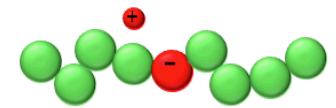
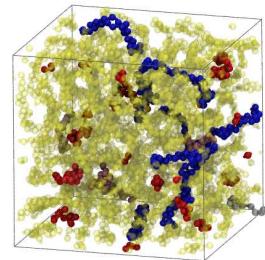
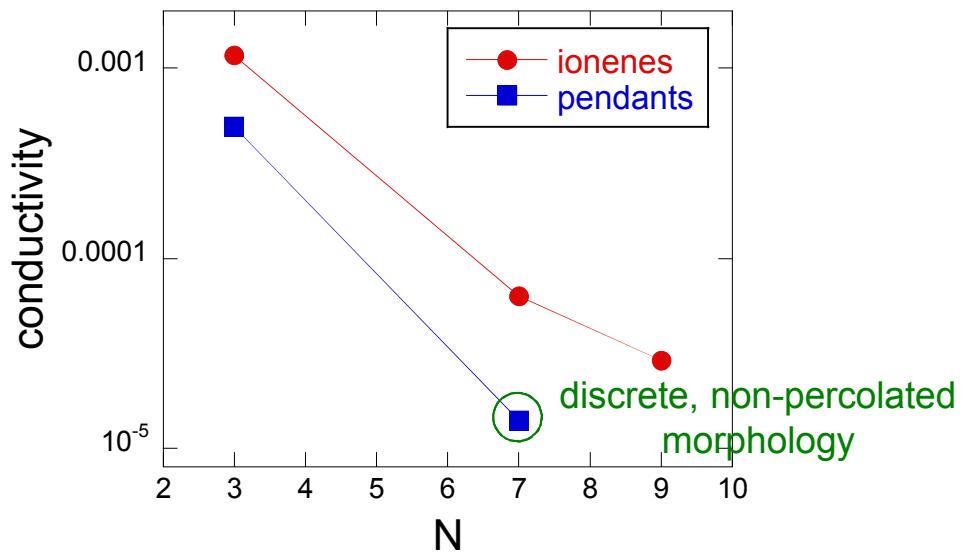
MD Structure factors





chains slow down early

Coarse-grained model



- conductivity decreases with decreasing ion concentration
- lowest for non-percolated aggregate morphology