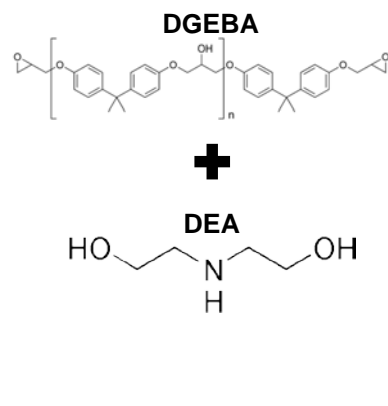


*Exceptional service in the national interest*

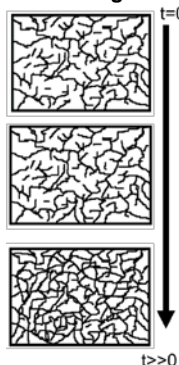


## Materials

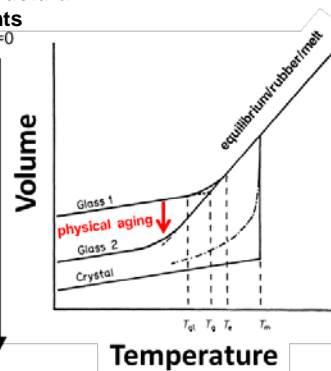


## Chemical and Physical Aging

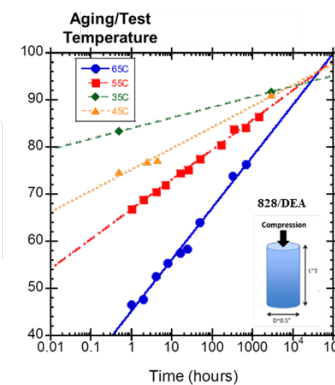
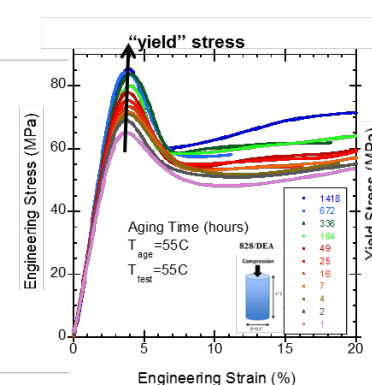
Additional Chemical Crosslinking



Physical or Structural Rearrangements



## Thermal-Mechanical Response Changes



**The interplay of chemical and physical aging in a diglycidylether of bisphenol A (DGEBA) epoxy cured with diethanolamine (DEA)**

Jamie M. Kropka, Gabriel K. Arechederra, and John D. McCoy  
Sandia National Laboratories, Albuquerque, NM

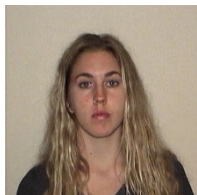
# Polymer Physics, Characterization, Modeling and Processing Group

## Experimentalists



*Materials  
Science &  
Engineering*

Kelsey Wilson



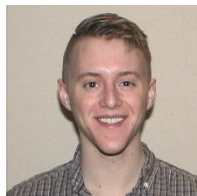
Lindsey Hughes



Doug Adolf (retired)



Taylor Gabaldon



Rex Jaramillo



Mark Stavig



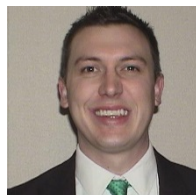
Nick Wyatt



Jamie Kropka



Cody Corbin



Mat Celina



John McCoy (NM Tech)



Jasmine Hoo (NM Tech)

Lara Draelos (NM Tech)

Maggie House (NM Tech)

Windy Ancipink (NM Tech)

## Modelers



Bob Chambers (retired)



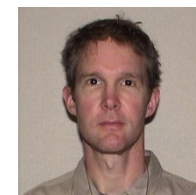
Kevin Long



Brenton Elisberg



Craig Tenney



Kurtis Ford



Matthew Neidigk (AFRL)



**Main Contributors to Today's Topics**



# Our Vision: Validated Model-Based Lifecycle Engineering for Packaging Design

## Polymer Nonlinear Viscoelastic (NLVE) Model

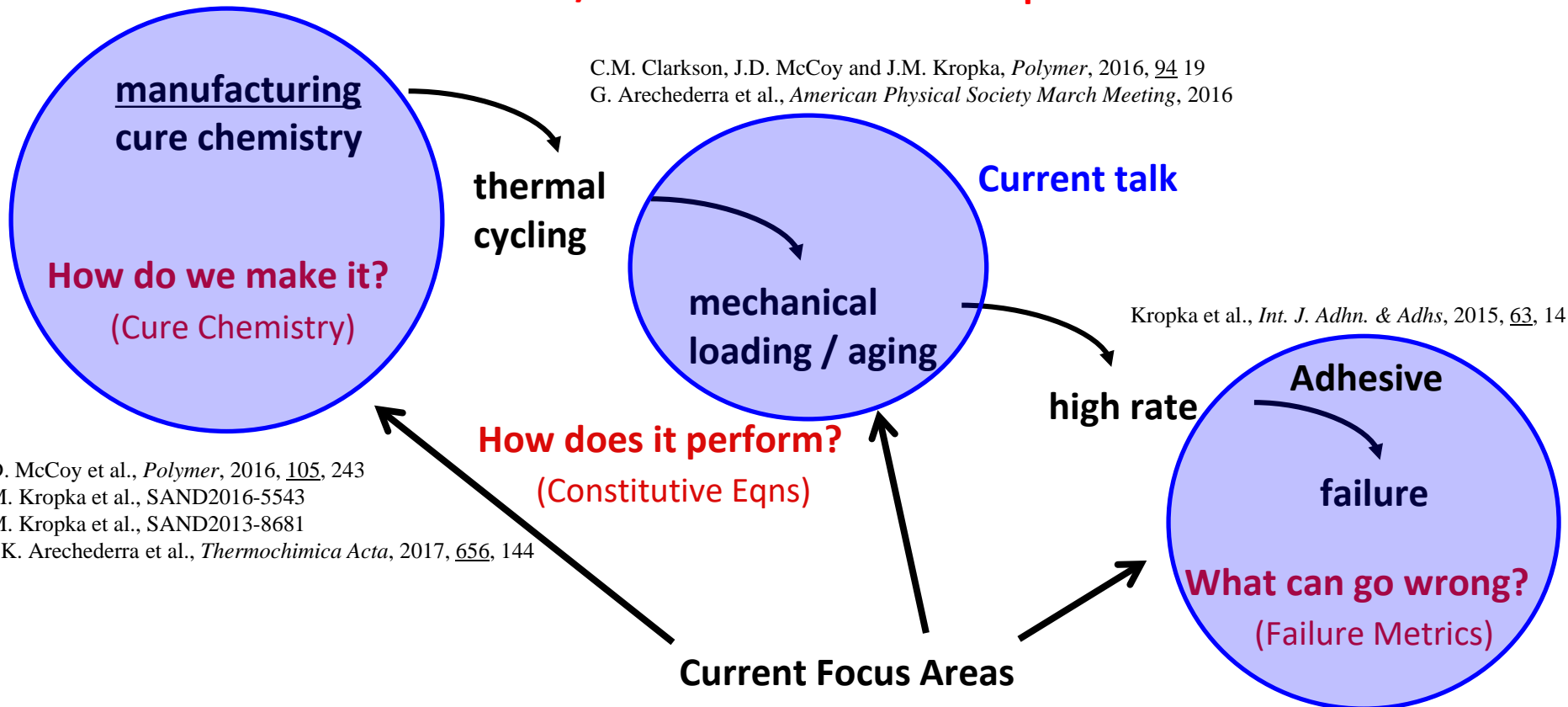
J.M. Caruthers, et al., *Polymer*, 2004, 45, 4577

D.B. Adolf, et al., *Polymer*, 2004, 45, 4599

D.B. Adolf, et al., *Polymer*, 2009, 50, 4257



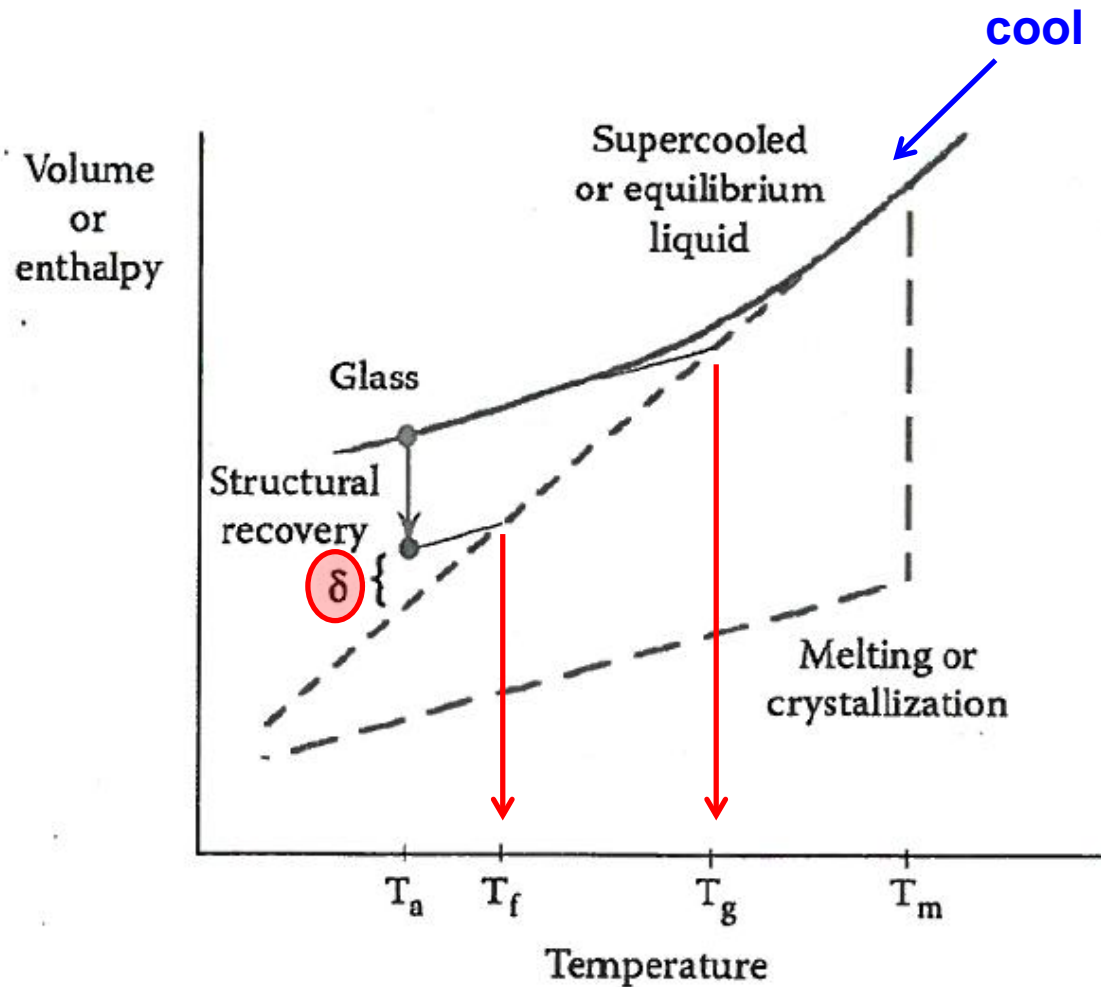
## Predict Stress/Strain and Understand Impact on Performance



# Polymer Glass Aging Topics

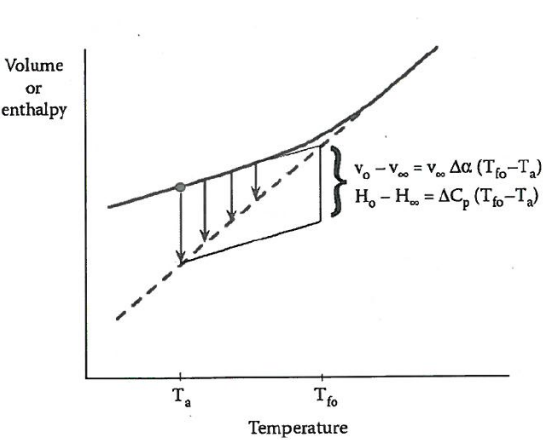
- Background
  - Glass Formation and Structural Recovery/Relaxation
  - Signatures and Impact of Structural Recovery/Relaxation
  - What is lacking in our understanding and what is left to do?
- Our Current Efforts
  - Goals
  - Materials
  - Volume and mechanical response changes associated with aging
  - Assessment of impact of aging on stress and failure in application relevant geometries

# Glass Formation and Structural Recovery/Relaxation

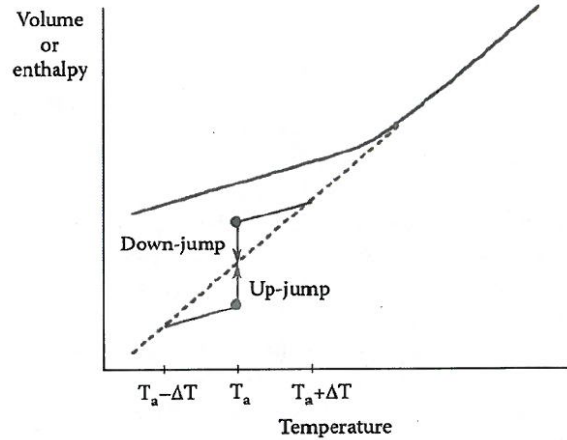


# Signatures of Structural Recovery/Relaxation

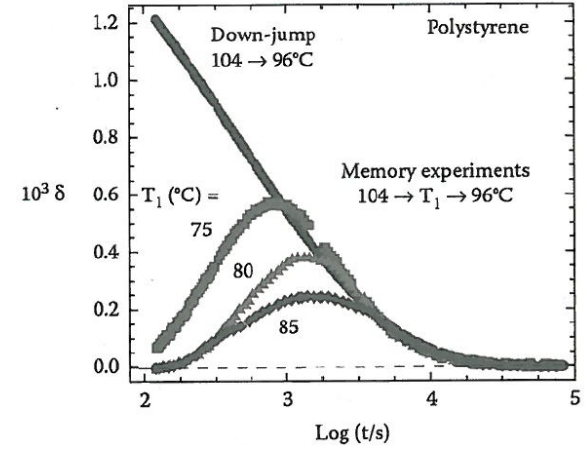
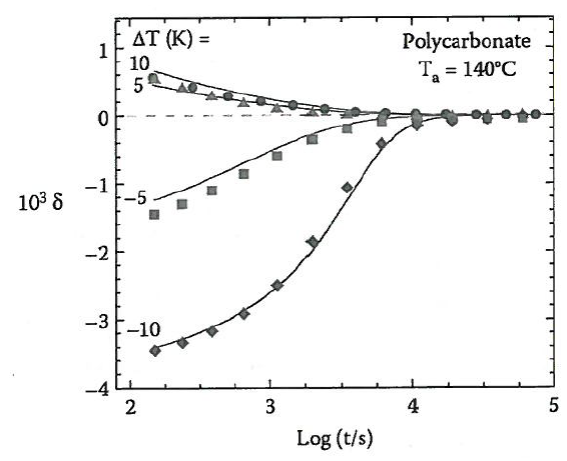
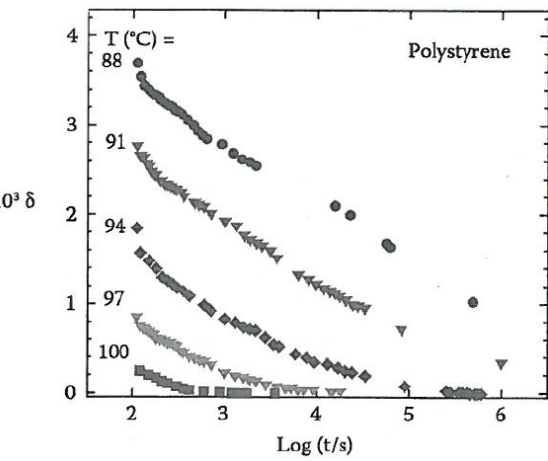
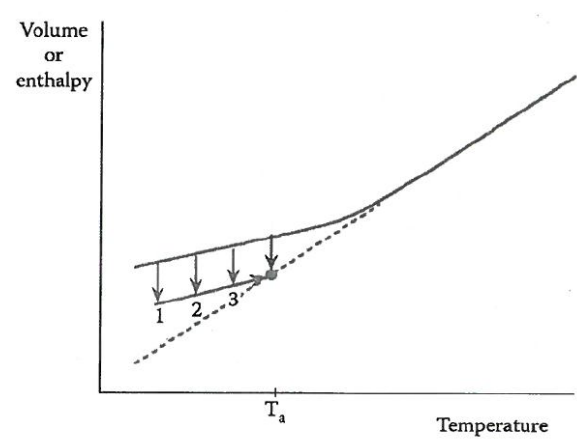
## Intrinsic Isotherms



## Asymmetry of Approach



## Memory Effect



Relaxation Depends on Structure

Relaxation Depends on History

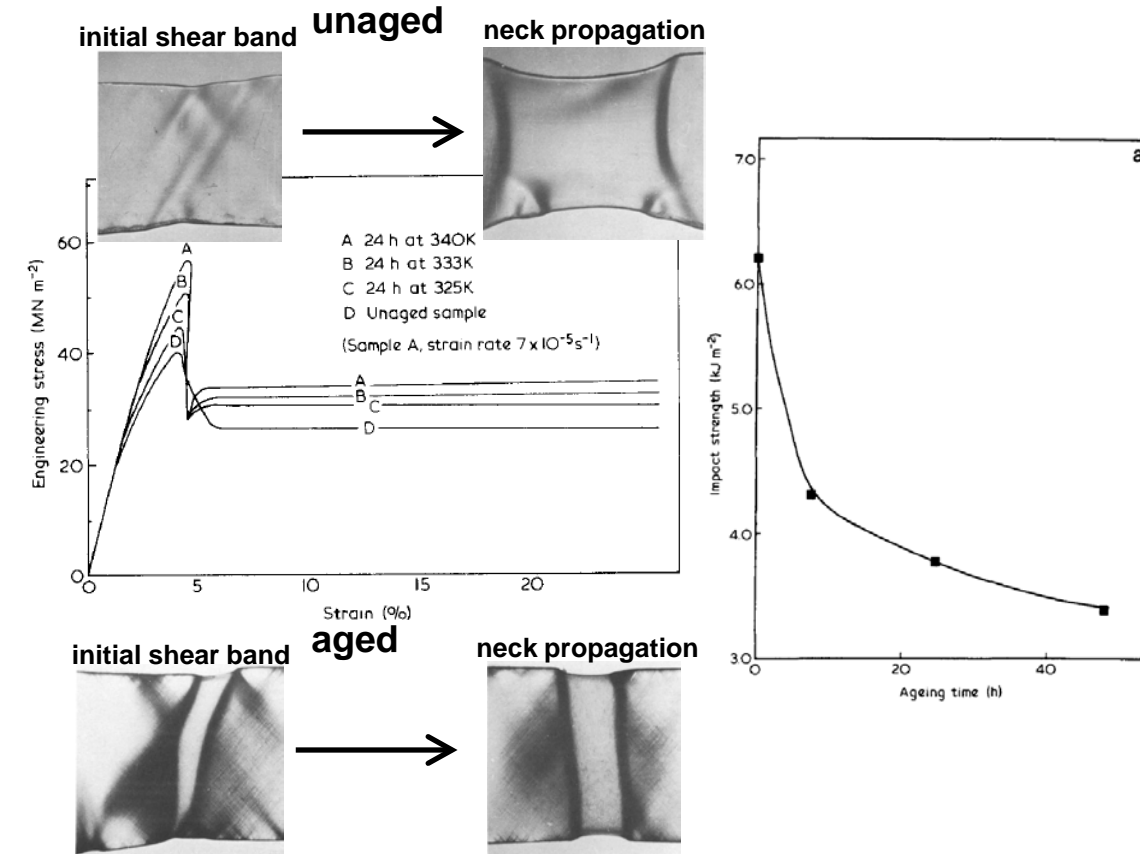
**KAHR and TNM models capture qualitative features of glassy kinetics and the 3 signatures of structural recovery**

# Impact of Structural Recovery and Physical Aging

“Failure modes of polymers can change from ductile to brittle failure with aging”

S.L. Simon and G.B. McKenna, in *Polymer Glasses*, 2017, pg. 46

## Tensile and impact tests of PET during isothermal “aging”



## Izod impact studies of PC during isothermal “aging”

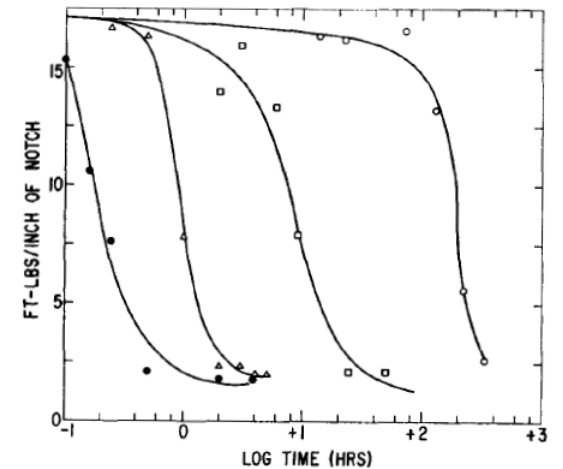


Fig. 3. Effect of annealing temperature on Izod impact data. ○) 100; □) 115; △) 125; ●) 130;  $[\eta] = 0.58$ .

D.G. Legrand, *J. Appl. Pol. Sci.*, 1969, **13** 2129

R.N. Haward et al., *Polymer*, 1983, **24** 1245

These are thermoplastics, but the phenomena can occur in thermosets too



# What is left to do?

“Further work and direct measurement of the volume and enthalpy along with the mechanical (physical aging) experiments should be undertaken on the same samples”

S.L. Simon and G.B. McKenna, in *Polymer Glasses*, 2017

- Currently probing epoxy volume/enthalpy relaxation plus changes in mechanical response AND using this information to design “strength” experiments in application relevant geometries

“...because the (KAHR and TNM) models do still exhibit some difficulties in quantitative prediction with model parameters showing a dependence on thermal history...” efforts are necessary to improve upon these models

S.L. Simon and G.B. McKenna, in *Polymer Glasses*, 2017

- Currently testing Sandia’s non-linear viscoelastic modeling capabilities against aging data

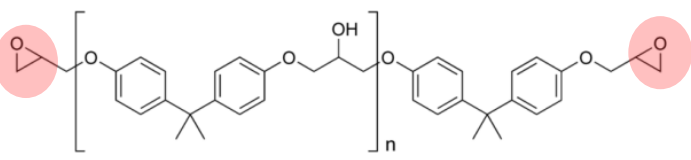
**Is physical aging a concern in terms of stress evolution in application designs?**



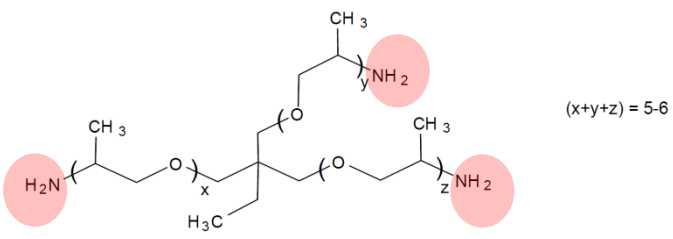
# Materials

## 828/T403<sup>1</sup> and 828/GMB/T403

EPON<sup>®</sup> Resin 828  
Diglycidylether of Bisphenol-A



Jeffamine<sup>®</sup> T-403 Polyetheramine

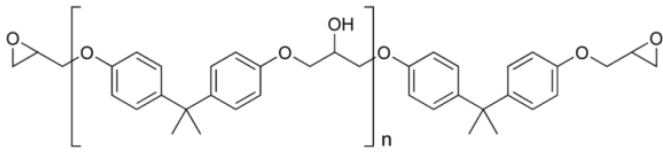


3M D32 glass microballoons

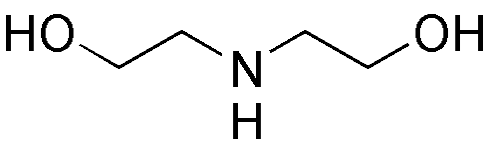
$T_g \sim 90C$   
(when mixed stoichiometrically epoxy-amine)

## 828/DEA<sup>2</sup> and 828/GMB/DEA<sup>3</sup>

EPON<sup>®</sup> Resin 828  
Diglycidylether of Bisphenol-A



Diethanolamine

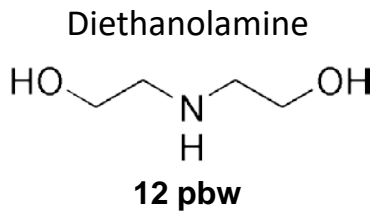
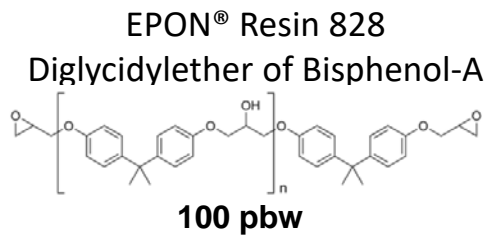


McCoy et al. *Polymer* **2016**, 105, 243-254.

$T_g \sim 70C$

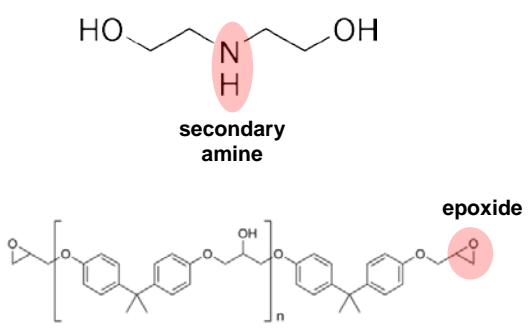
<sup>1</sup>Mix ratio, cure schedule, and more can be found in SAND2013-8681  
<sup>2</sup>Mix ratio, cure and typical properties can be found at: [http://www.sandia.gov/polymer-properties/828\\_DEA.html](http://www.sandia.gov/polymer-properties/828_DEA.html)  
<sup>3</sup>Mix ratio, cure and typical properties can be found at: [http://www.sandia.gov/polymer-properties/828\\_DEA\\_GMB.html](http://www.sandia.gov/polymer-properties/828_DEA_GMB.html)

# 828/DEA<sup>1</sup>



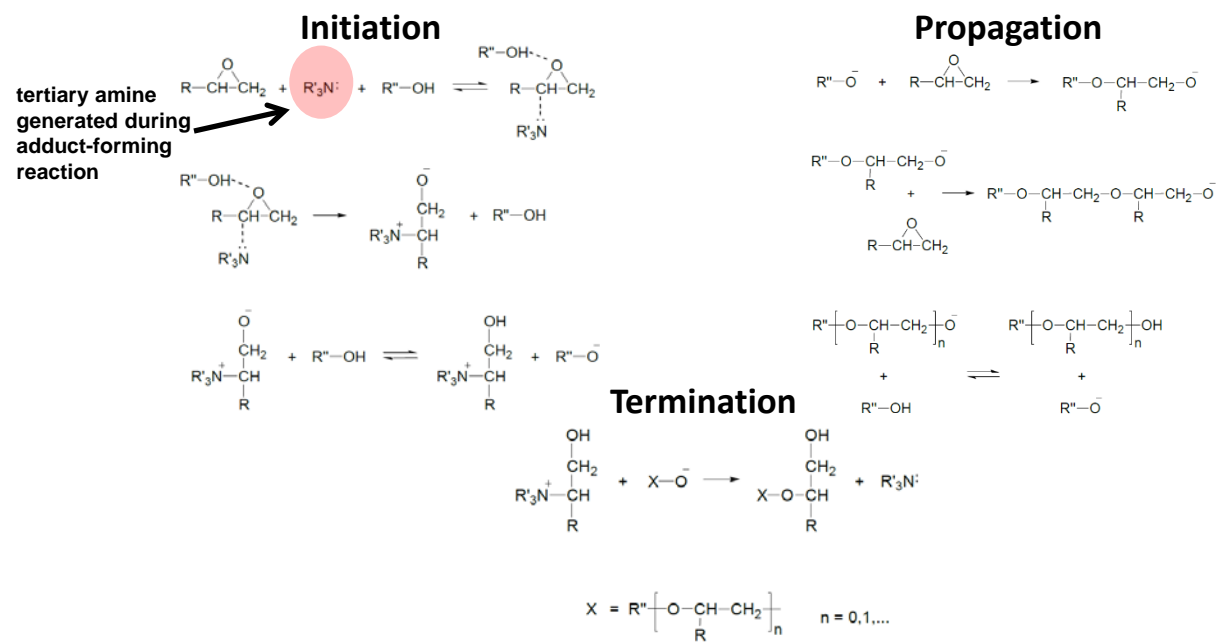
## Polymerization at T = 70°C (the cure process before aging)

### Adduct-Forming Reaction



All secondary amine is consumed in an addition reaction and excess epoxide remains

### Proposed Gelation Reaction



Anionic Chain-Growth Polymerization Catalyzed by Tertiary Amine from Adduct-Forming Reaction

J.D. McCoy et al., *Polymer*, 2016, 105, 243

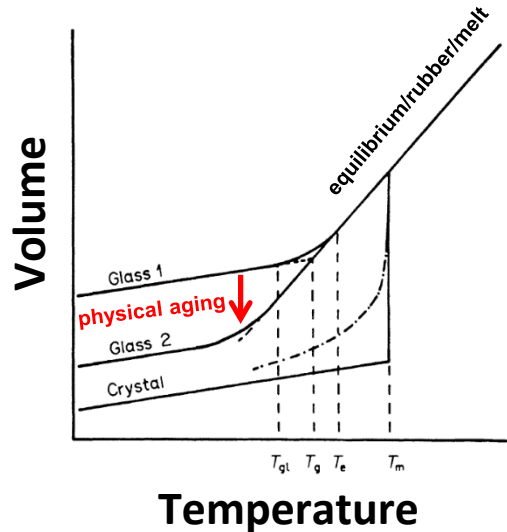
$T_g \sim 70^\circ\text{C}$

[when mixed 100:12 (pbw) 828:DEA and cured 24 hours at T=70°C ]

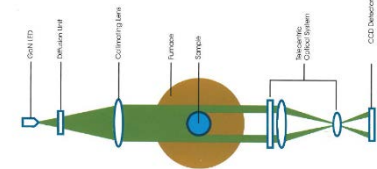
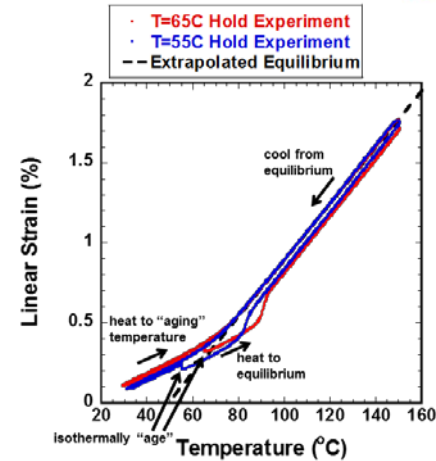
<sup>1</sup>Mix ratio, cure and typical properties can be found at: [http://www.sandia.gov/polymer-properties/828\\_DEA.html](http://www.sandia.gov/polymer-properties/828_DEA.html)

# Polymer Glass Aging

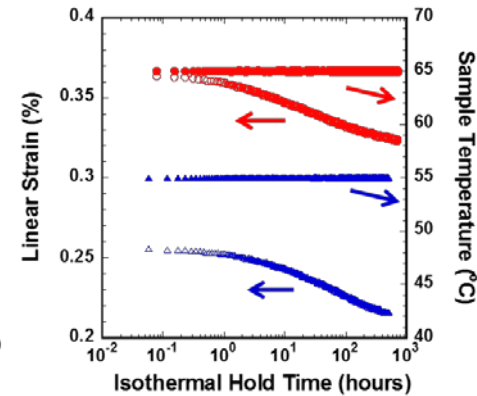
## Material Volume Changes



optical resolution

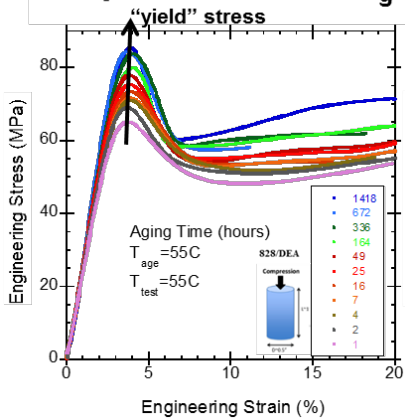


Isothermal Hold Response

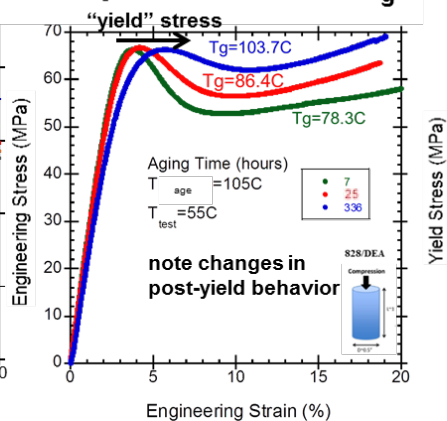


## Material Mechanical Response Changes

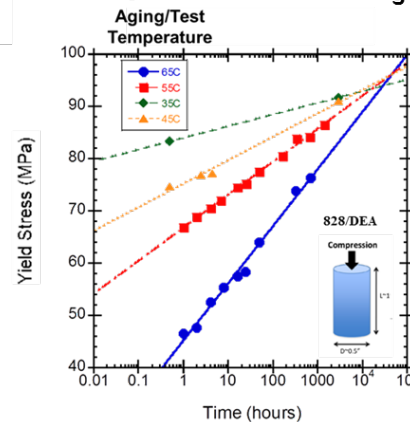
Exposure Below  $T_g$



Exposure Above  $T_g$



Exposure Below  $T_g$

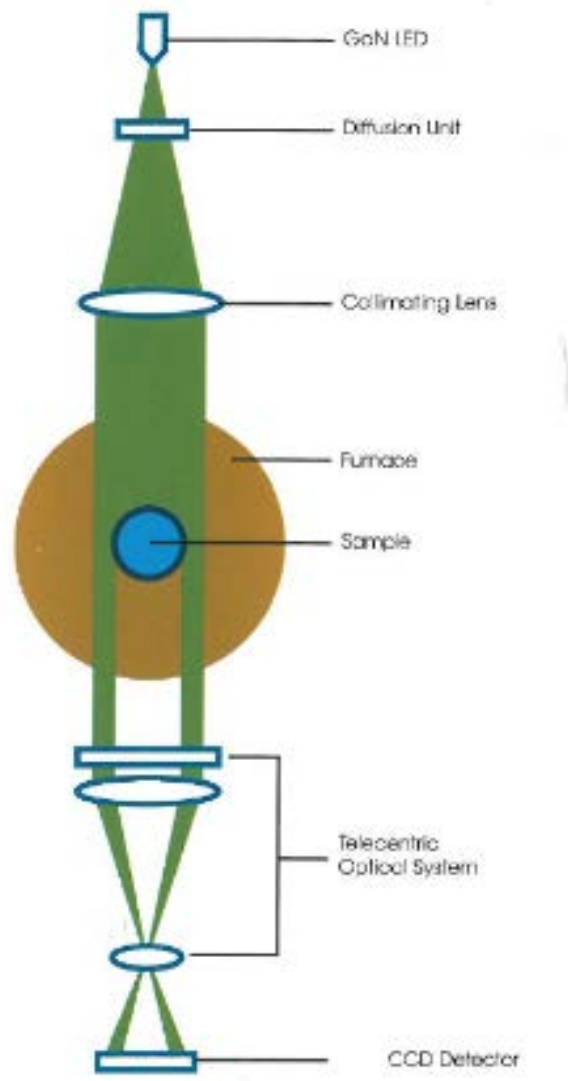


SNL NLVE polymer models (e.g., SPEC) have the framework to predict the aging behavior and should be tested against measurements

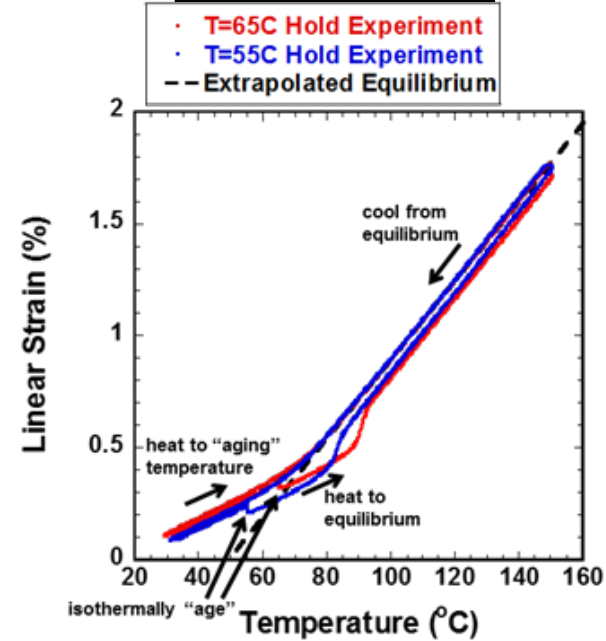
**Volume**

# Measuring Volume Response Associated with Aging

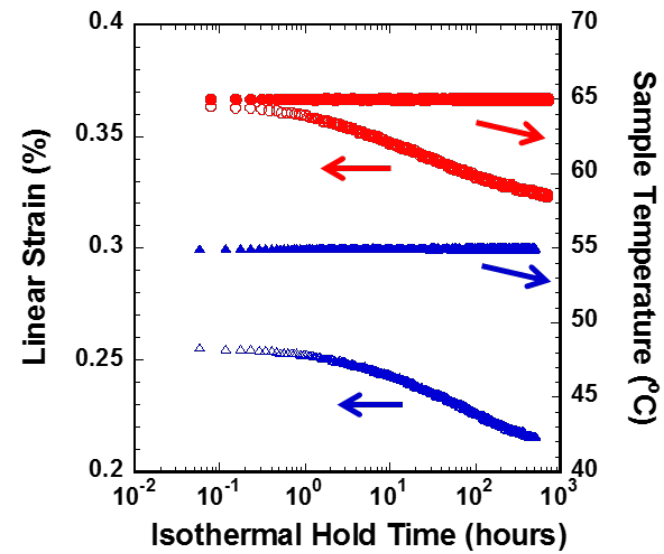
## Optical Resolution of Length\*



## Full Experiment

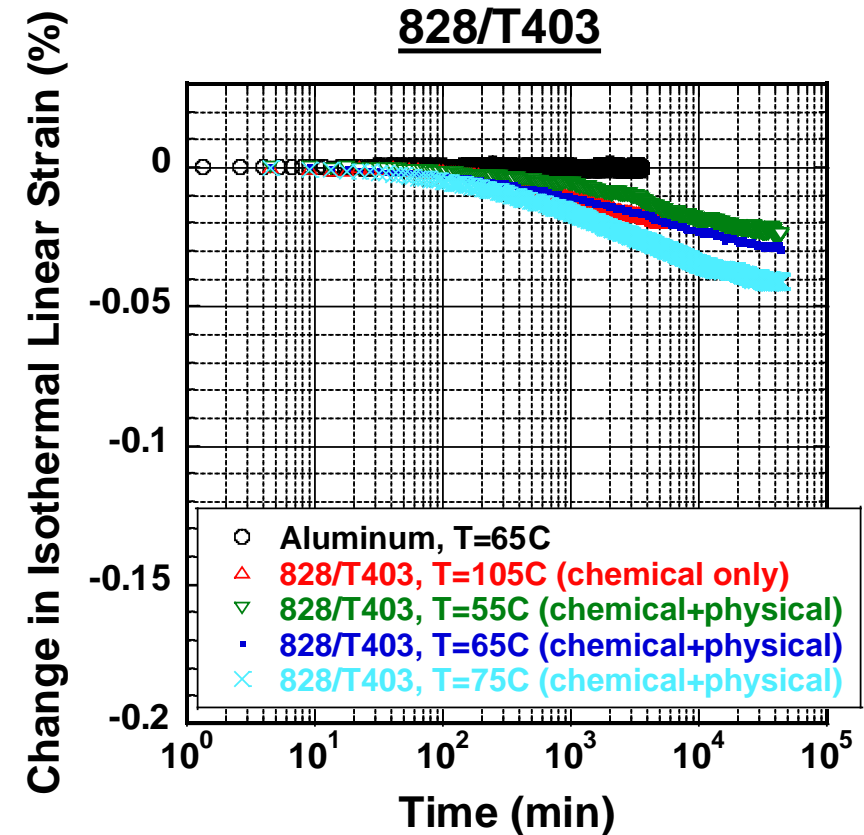
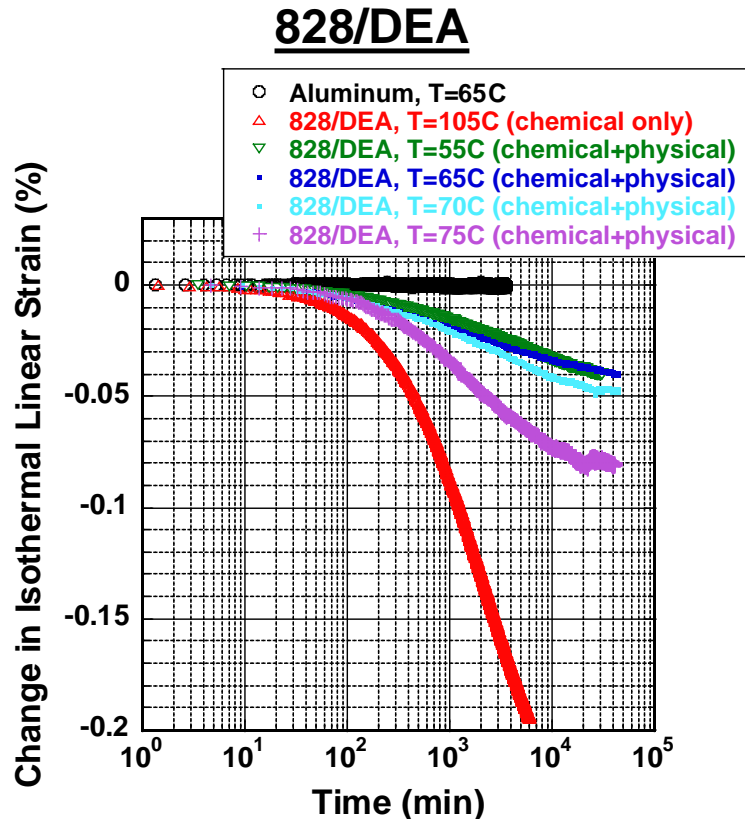


## Isothermal Hold Response



\*for isotropic materials  $\Delta V = 3\Delta L$

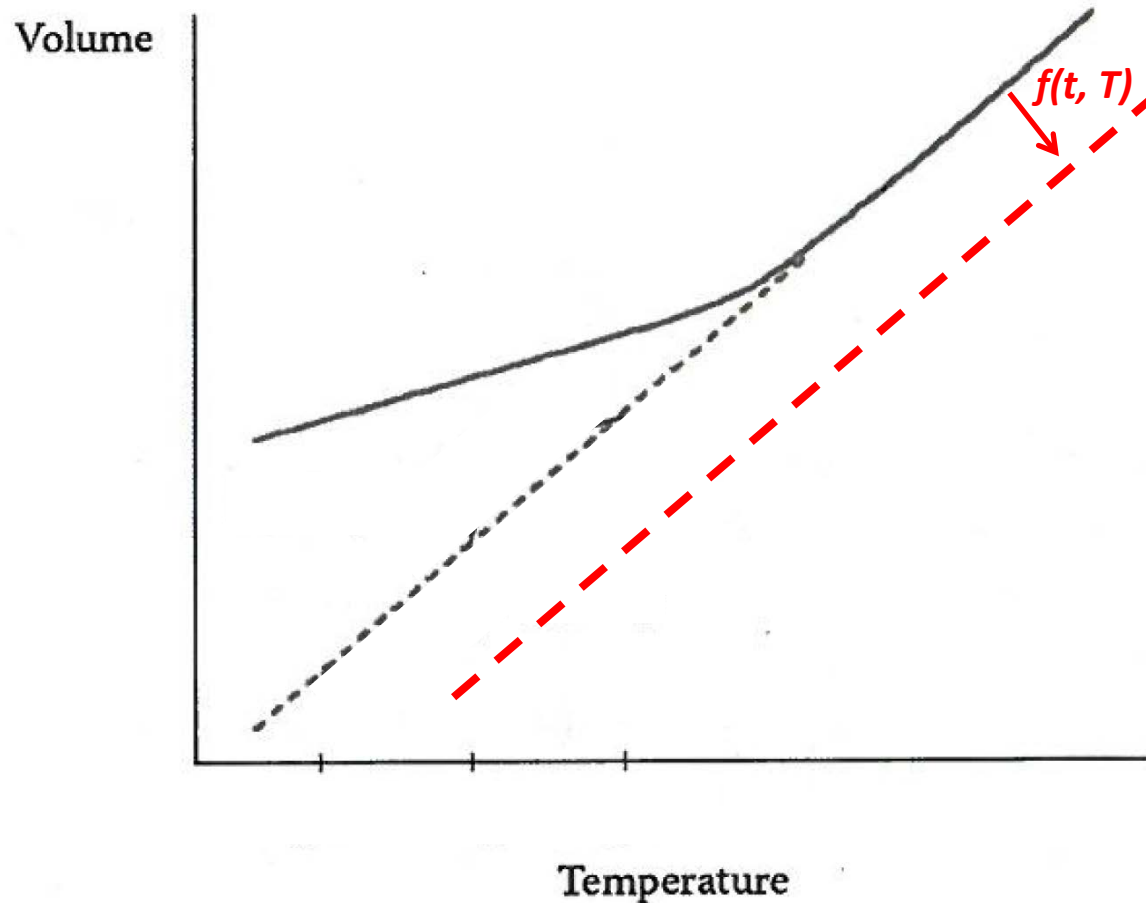
# Isothermal Volume Response for 2 Common Epoxy Thermosets



**Note:** Remaining reactive potential (excess epoxide groups in the case of 828/DEA) can play a significant role in total volume change

- The 50 nm instrument (length) resolution enables quantitative tracking of material length over time and provides the opportunity to resolve functionality [e.g.,  $l(t)$ ] that describes material behavior
- Minimizing potential for continued cure during “aging” by using “stoichiometric” epoxy thermosets (e.g., 828/T403) can have significant impact on material “shrinkage” magnitude

# Challenges in Defining Equilibrium for 828/DEA

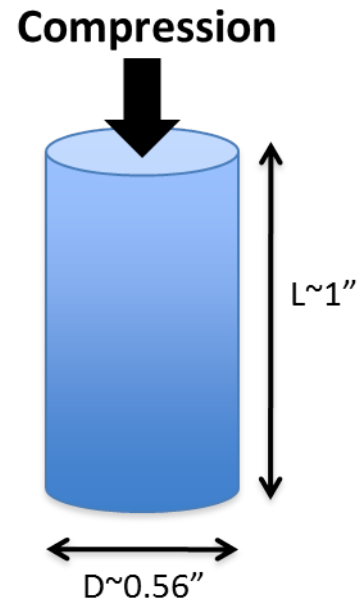
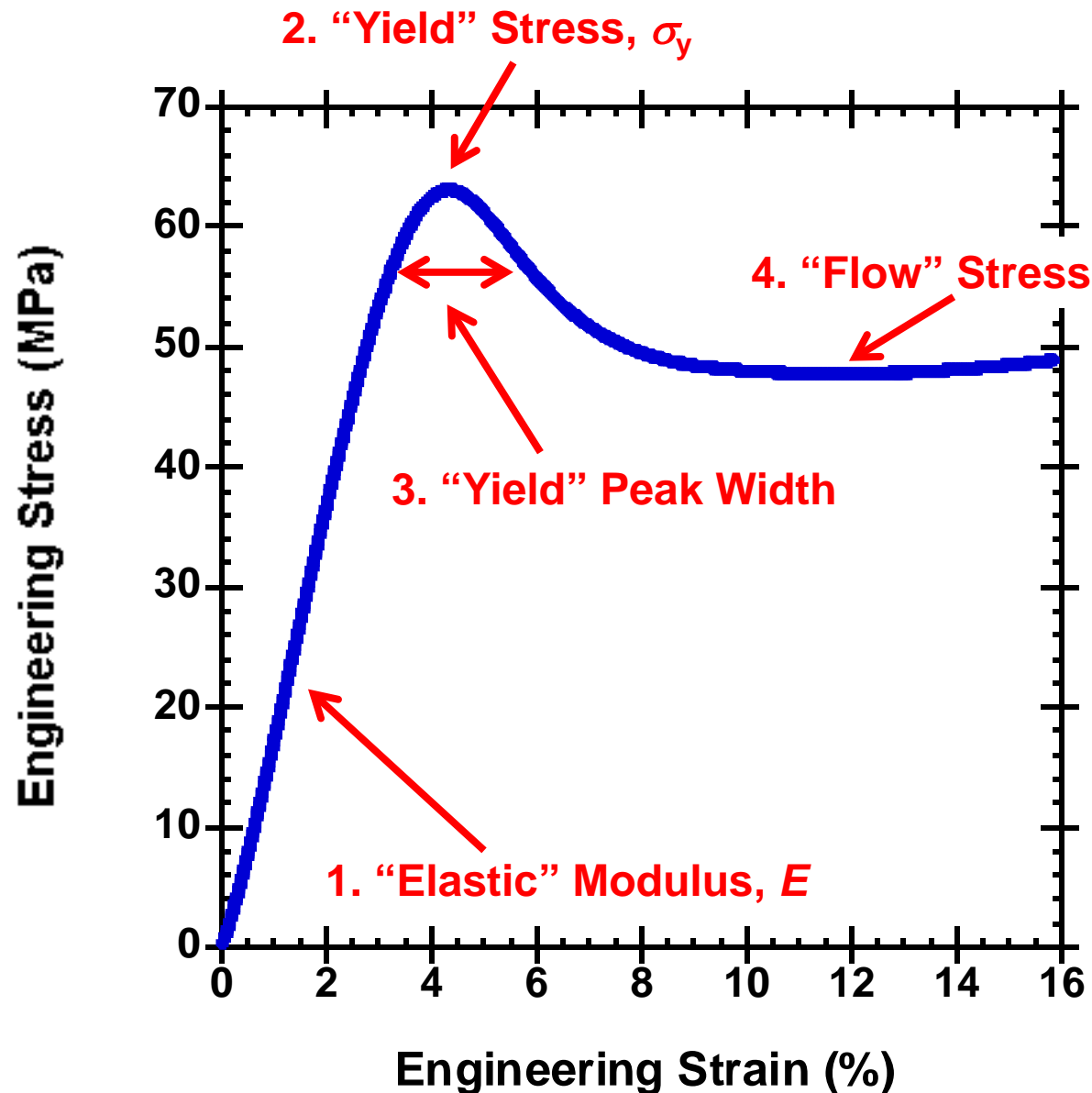


The rate of change of the equilibrium line (associated with continued epoxide cross-linking in 828/DEA) depends on the time and temperature history of the material and can only be determined for a particular history by re-equilibrating the material

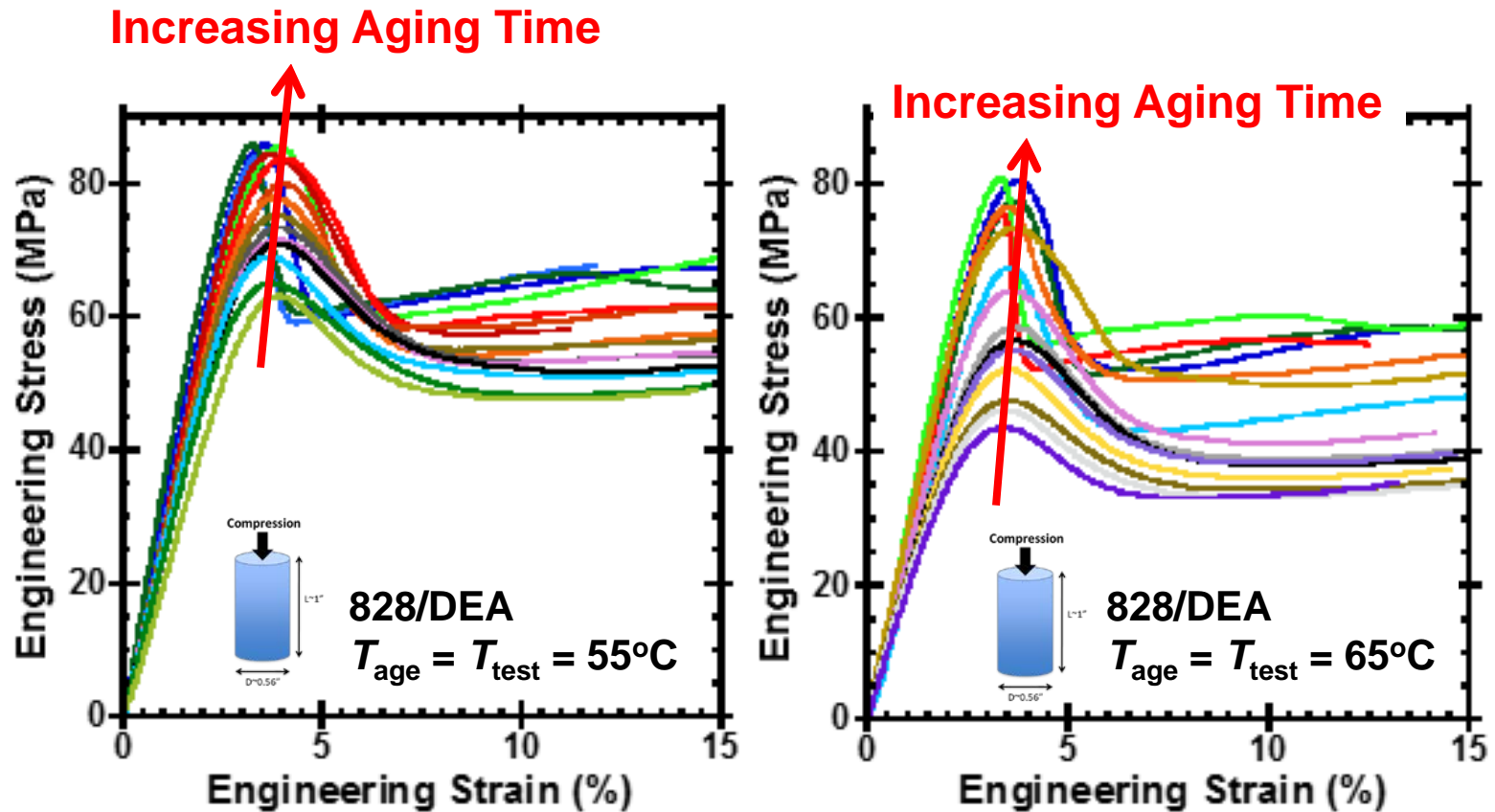


**Mechanical**

# Anatomy of Compressive Stress-Strain Response of Glassy Polymers



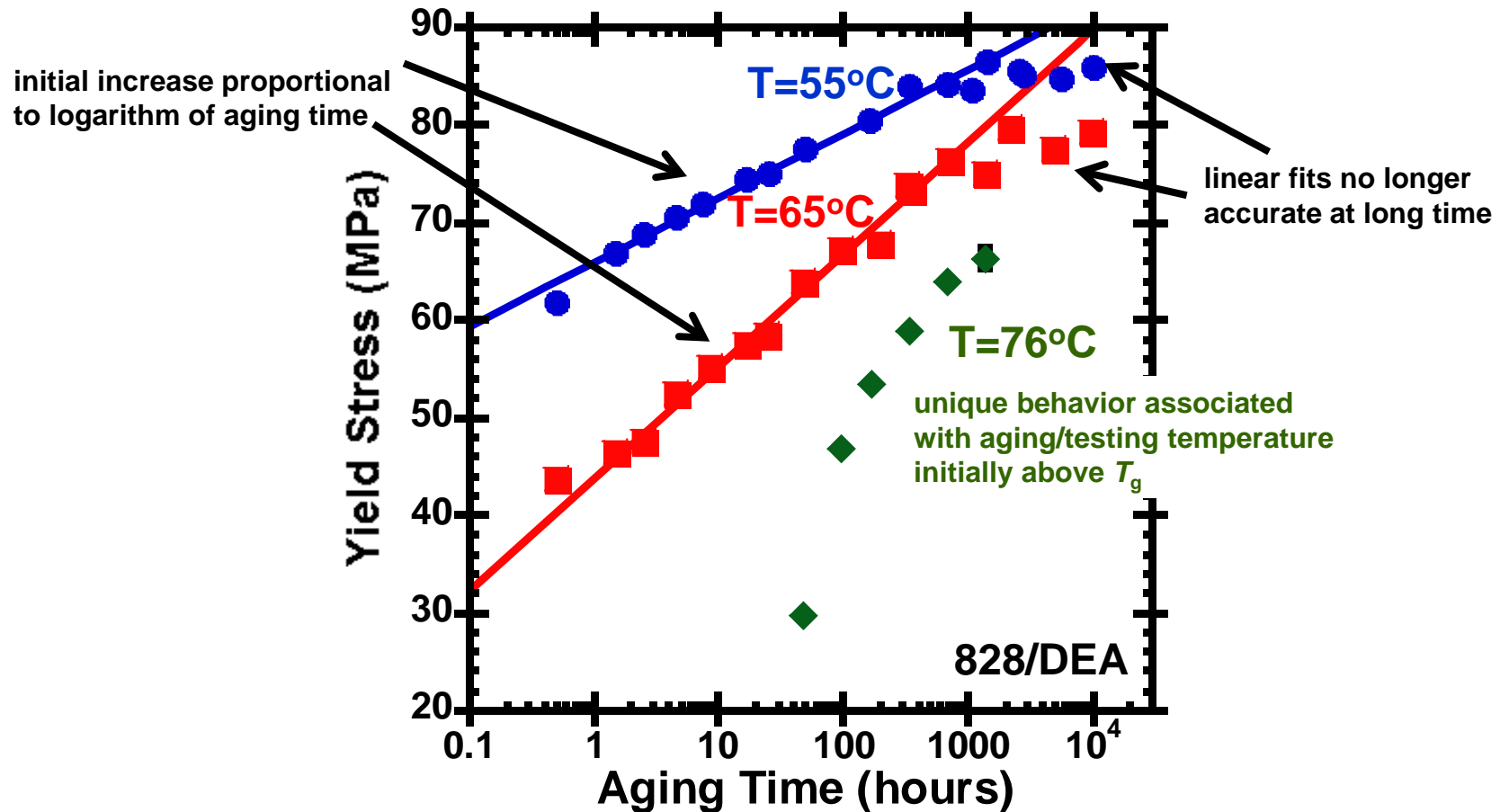
# Changes in Compressive Stress-Strain Response Associated with Thermal Aging



## 4 Distinguishable Changes in Compressive Stress-Strain Response Include:

- Increase in “elastic” compressive modulus
- **Increase in “yield” stress**
- Narrowing of “yield” peak
- Increase in “flow” stress

# Evolution of Yield Stress during Thermal Aging



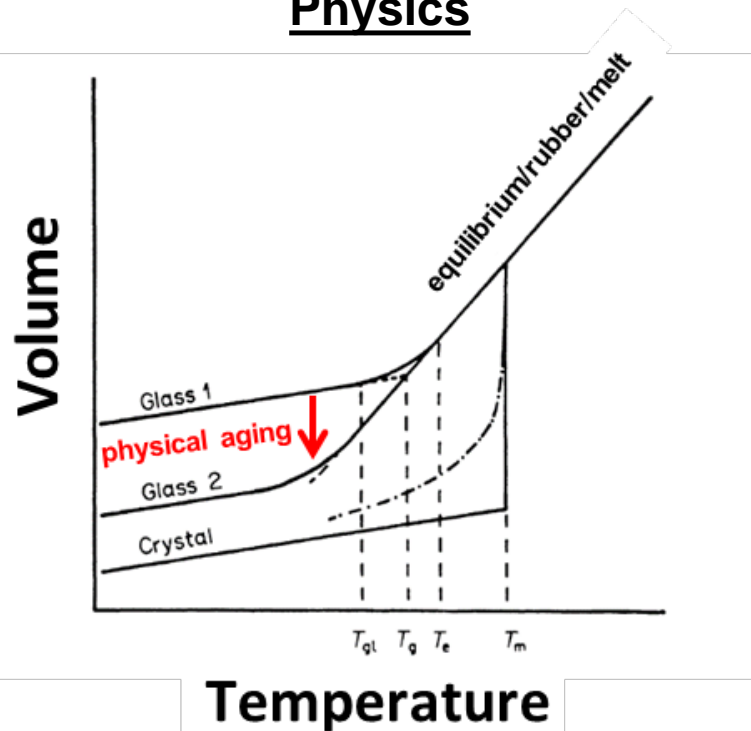
Focusing on  $T = 55^{\circ}\text{C}$  and  $65^{\circ}\text{C}$  datasets for now:

- Changes in yield stress are substantial—as high as 82%
- The evolution of yield stress with time changes (or possibly stops) after ~30 days

**What is the mechanism(s) driving this change?**

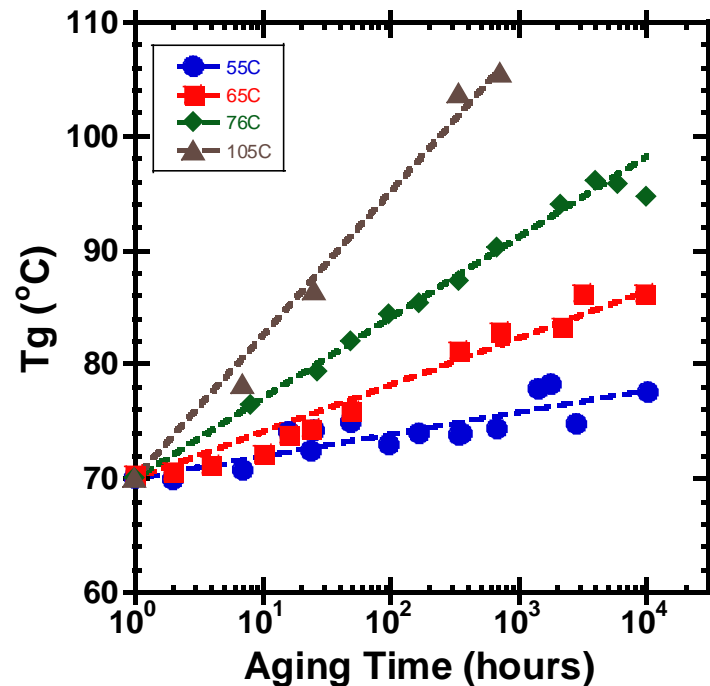
# Mechanisms Driving Evolution of Yield Stress during Thermal Aging

## Physics



Volume relaxation (densification) of the material slows molecular motions in the polymer chain and this contributes to an increase in the observed yield stress in the compressive stress-strain response

## Chemistry



Continued chemical crosslinking increases the glass transition temperature of the material. This also slows molecular motions in the polymer chain (at a given temperature below  $T_g$ ) and contributes to an increase in the observed yield stress in the compressive stress-strain response

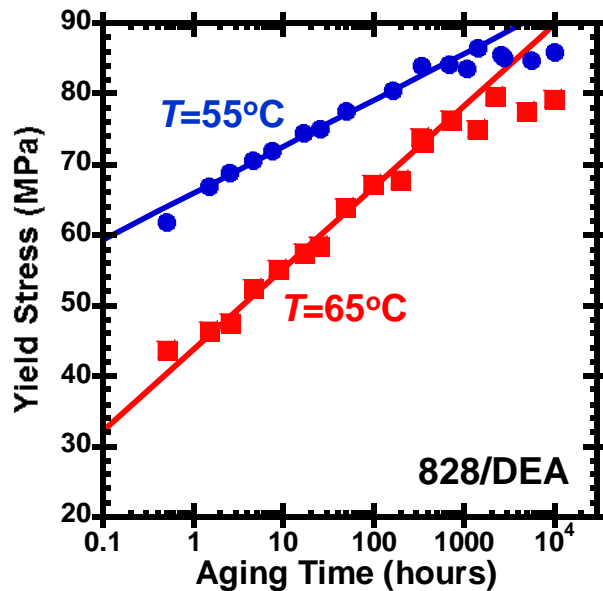
**Can these contributions to the overall increase in yield stress be separated?**

# Chemical and Physical Contributions to the Evolution of Yield Stress during Thermal Aging

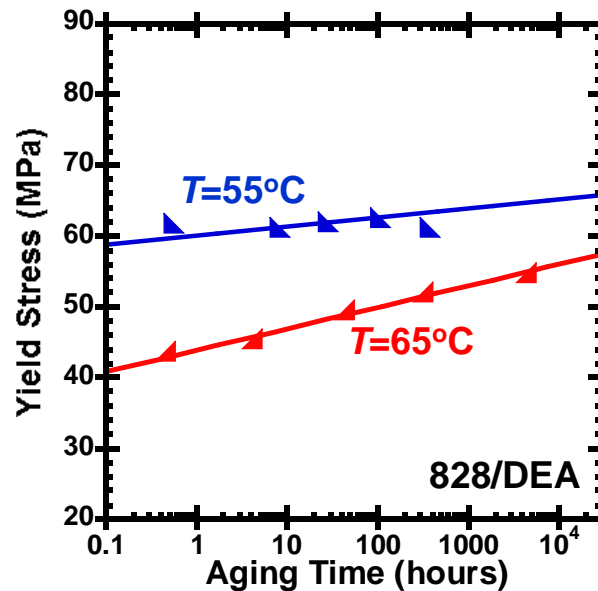
Anneal → Age → Test

Anneal → Age → Anneal → Test

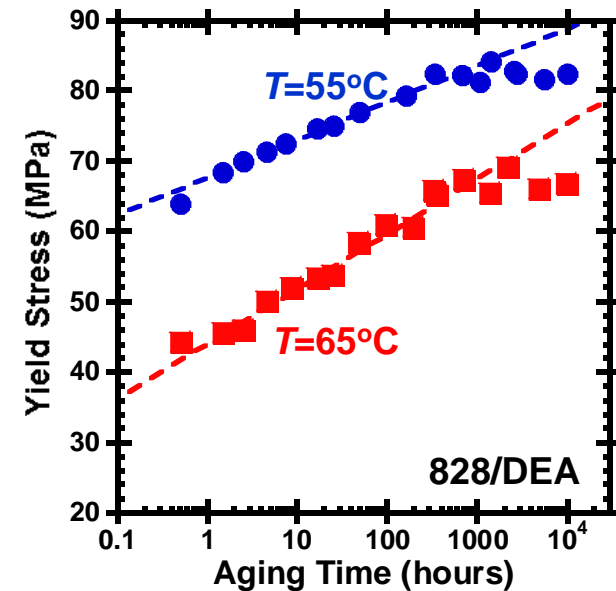
Chemical + Physical  
(Measured)



Chemical Only  
(Measured)

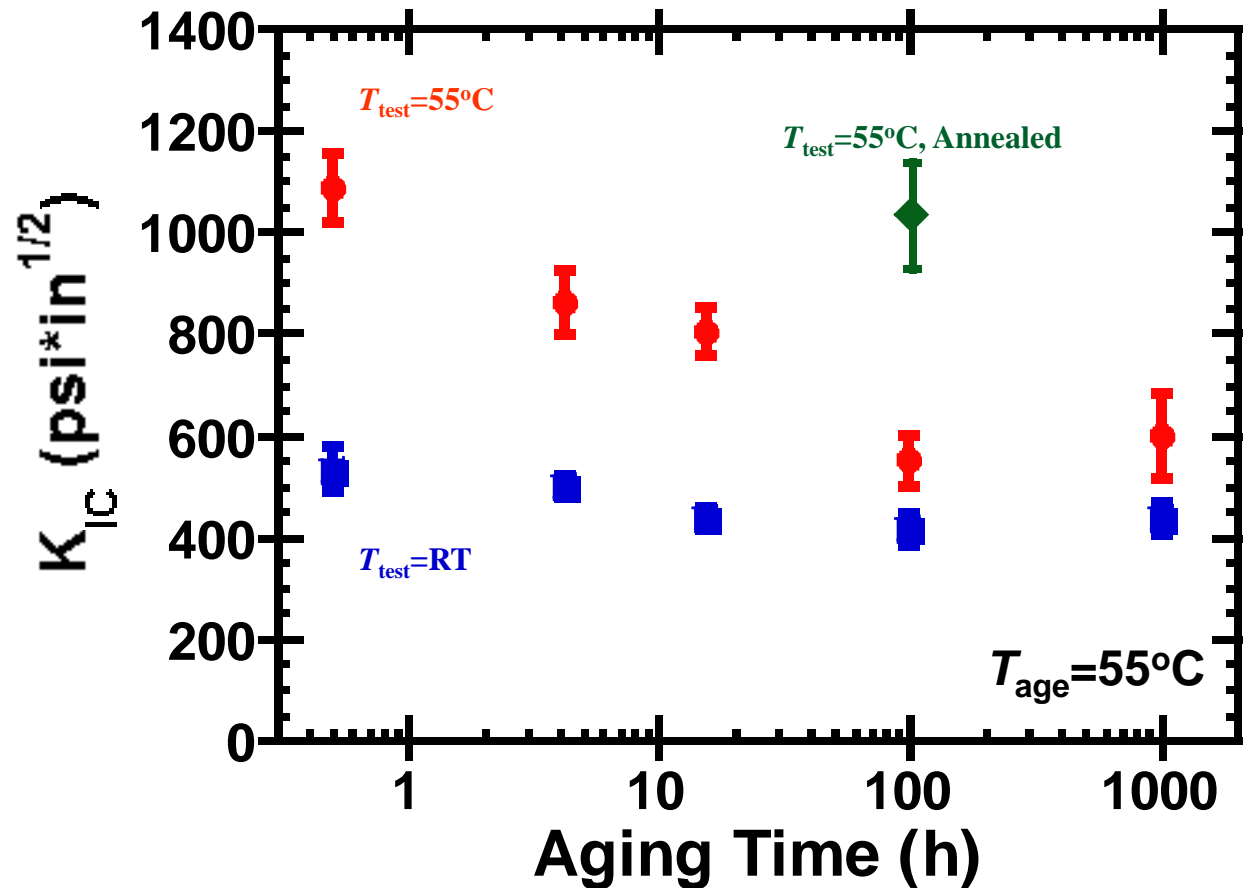


Physical Only  
(Calculated)



By thermally annealing the samples above the glass transition temperature (after aging), the physical history of the sample is erased and the chemical-only contributions to the evolution of the yield stress are resolved. Physical-only contributions are calculated by subtracting the chemical-only contributions from the total change in yield stress.

# Fracture Toughness Changes with Aging Too



Fracture Toughness Changes Occur Over the Same Timescale  
and are Associated with Structural Relaxation

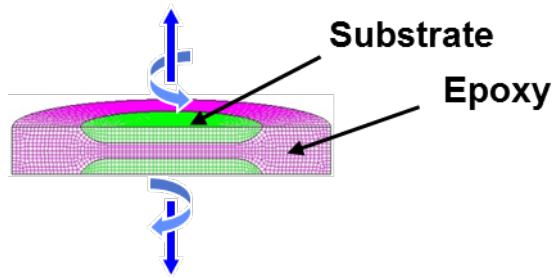
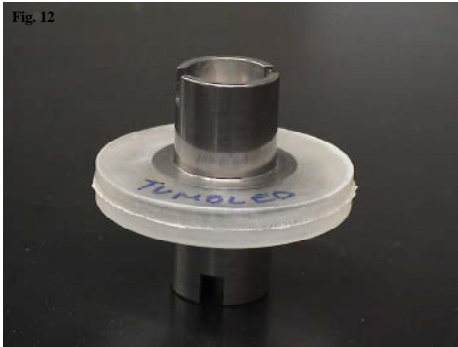


# Summary

- Demonstrated ability to resolve in-situ material dimensional changes associated with isothermal aging under no mechanical load
- Illustrated differences in dimensional changes between materials associated with the specifics of a given material (e.g., remaining reaction potential that can occur under the aging conditions)
- Resolved substantial changes in the compressive yield stress (as high as 80%) of the 828/DEA material over relatively short times (~30 days) when aged and tested below, but near, the glass transition temperature (e.g.,  $T_g-10^\circ\text{C}$ ,  $T_g-20^\circ\text{C}$ )
- Resolved the apparent attainment of equilibrium, at which time there is no further change (associated with physics) in yield stress
- Discriminated between the chemical and physical contributions to the evolution of the yield stress during isothermal aging

# Impact of Aging in Application Relevant Geometries

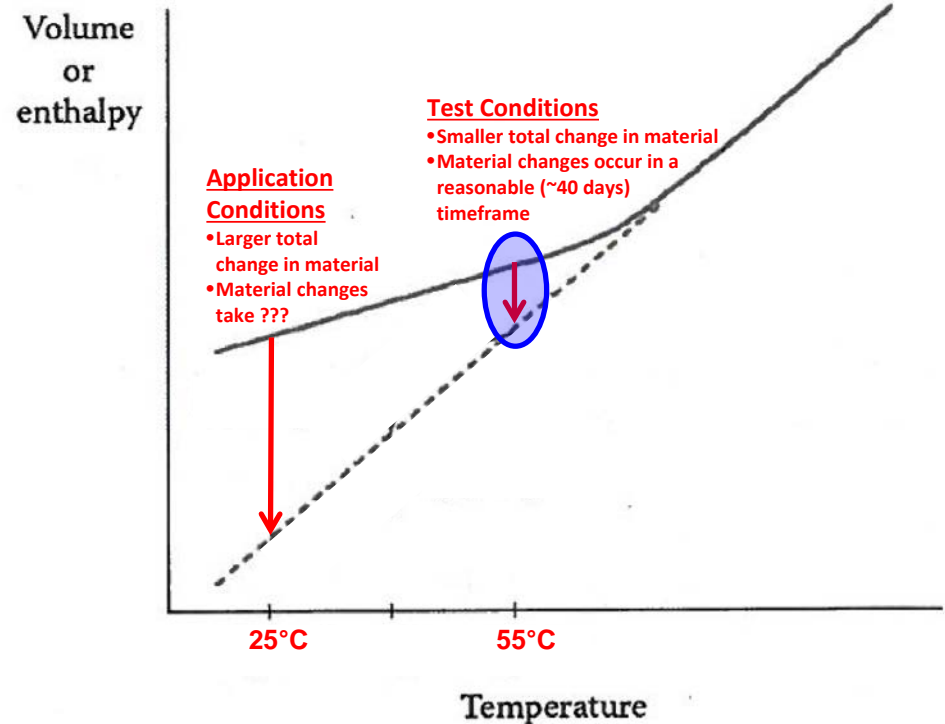
## Saucer Test Geometry



- Max stresses do not reside at an air interface (failure at “embedded interface”)
- Max stresses are smooth functions, not “spiked”

Initial focus on tensile loading only  
(it may be the most sensitive to aging)

## Aging Conditions



Results Coming Soon

# Final Remarks

- We are actively examining structural recovery (volume, enthalpy) and physical aging (e.g., compressive stress-strain, fracture toughness) together in epoxy thermosets
  - Dimensional changes monitored at a high resolution
  - Significant changes in mechanical response (yield stress, fracture toughness) are observed to accompany structural relaxation
- Based on what is learned from materials testing, we are designing structural tests to examine the impact of materials aging on application designs
- More work is necessary to assess predictive capabilities of materials aging in order to build confidence in the tools to examine the impacts of application designs and environments