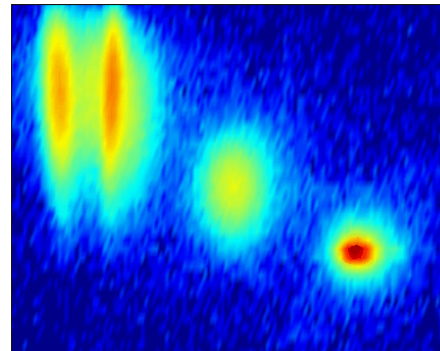
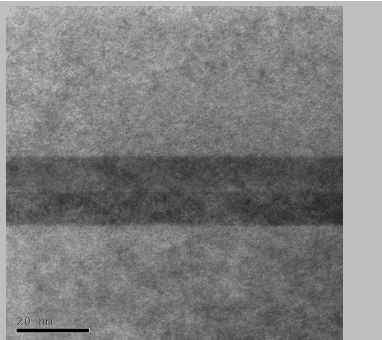
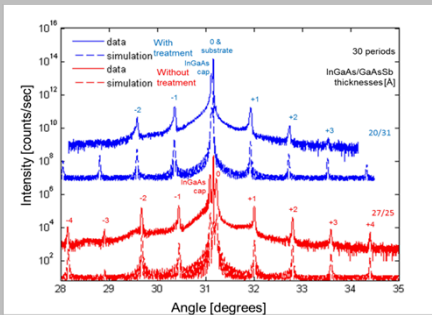


*Exceptional service in the national interest*



# Efforts to improve interfaces during growth of GaAsSb/InGaAs heterostructures

Jeff Cederberg, Eric Shaner, Emil Kadlec

*Sandia National Laboratory, Albuquerque, NM 87185 USA*

Alex Albrecht

*University of New Mexico, Physics and Astronomy, Albuquerque, NM 87131 USA*



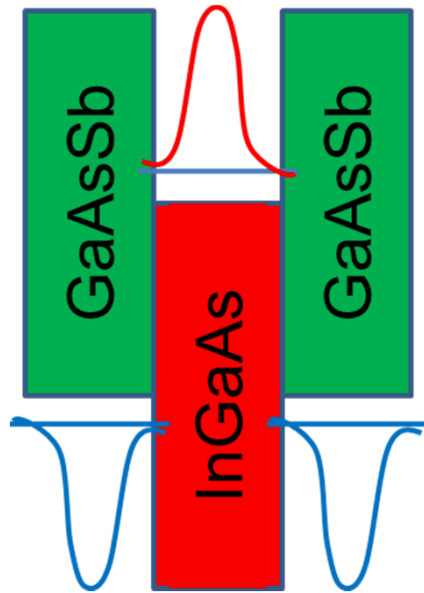
Sandia National Laboratories is a multi-program laboratory managed and operated by Sandia Corporation, a wholly owned subsidiary of Lockheed Martin Corporation, for the U.S. Department of Energy's National Nuclear Security Administration under contract DE-AC04-94AL85000.

# Why are we interested in GaAsSb/InGaAs heterostructures?



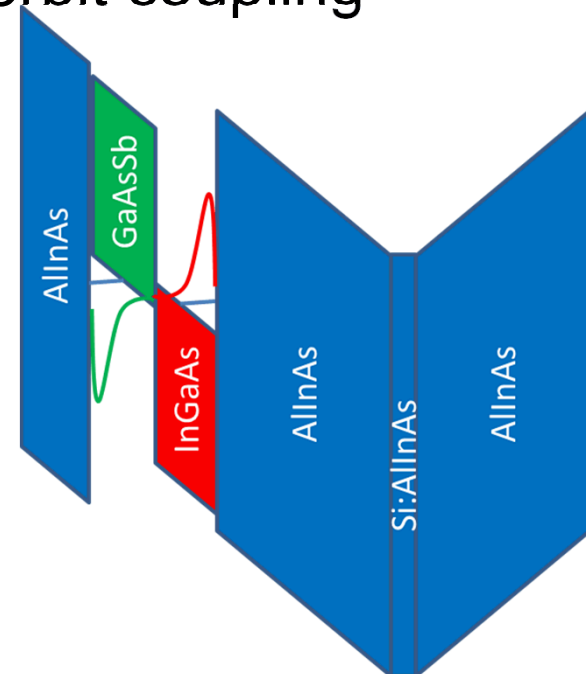
## Type II staggered offset

- Allows energies less than constituent band gaps to be achieved



## Large spin orbit energy

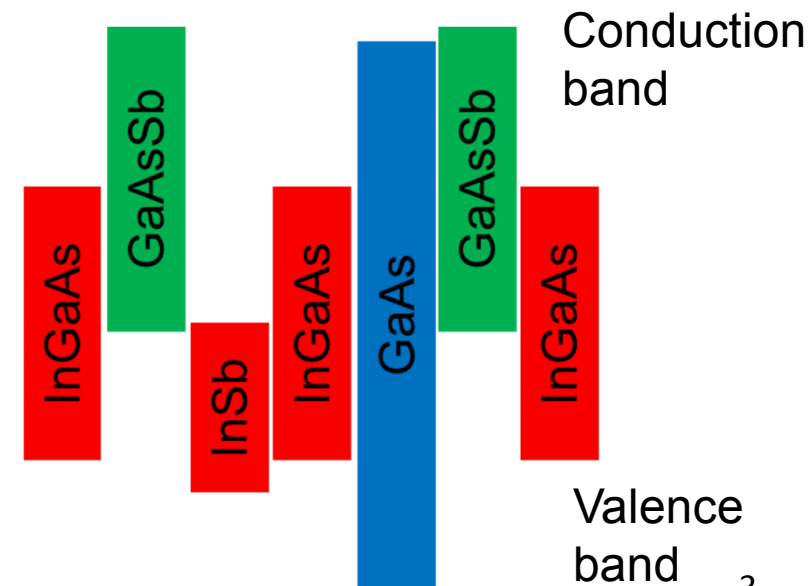
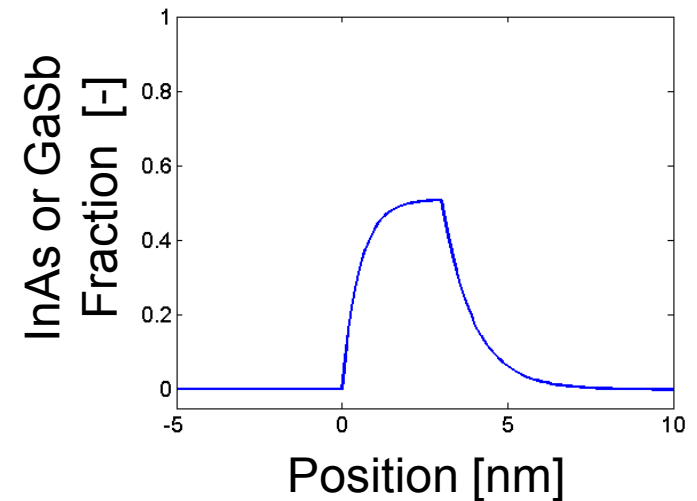
- Potential for electron spin polarization through spin-orbit coupling



# What is the challenge? Achieving abrupt profiles



- In and Sb are known to have slower response for both accumulation and desorption
- Imposed on this is the chemistry of the deposition process
  - Temperature drives source molecule decomposition
- Results in compressive InSb-rich and tensile GaAs-rich interface layers can dominate properties





# Outline

- Experimental details
- Ideal heterostructure characteristics
  - X-ray diffraction simulations
- Non-ideal interfaces
  - X-ray diffraction results
- InGaAs going to GaAsSb
- GaAsSb going to InGaAs
- Conclusions

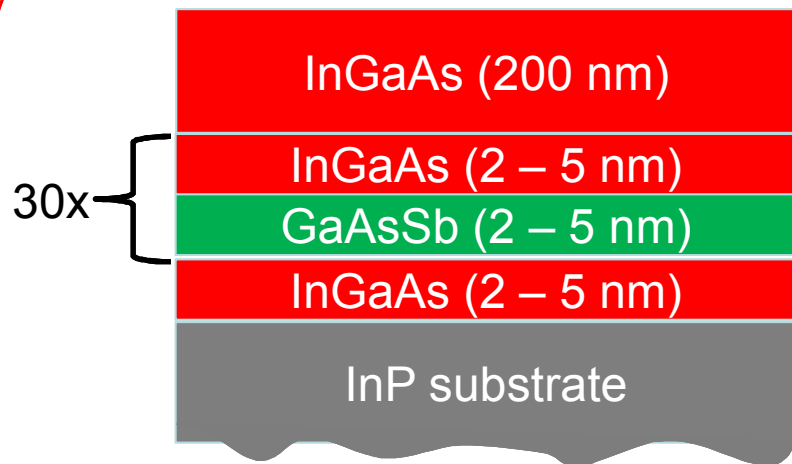
# Growth details

- Grow structure using 60 Torr MOVPE in  $H_2$
- InP buffer: TMIn and  $PH_3$  at  $620^\circ C$ 
  - Growth rate = 12 nm/min with V/III = 150
- AlInAs: TMIn, TMAI, and  $AsH_3$ 
  - Growth rate = 24 nm/min with V/III = 70 at  $660^\circ C$
- Temperature = 530 to  $570^\circ C$
- InGaAs: TMGa, TMIn,  $AsH_3$ 
  - Growth rate = 22 nm/min with V/III = 92
- GaAsSb: TEGa, TMSb, and  $AsH_3$ 
  - Growth rate = 12 nm/min with V/III = 7, TMSb/ $AsH_3$  = 3.7



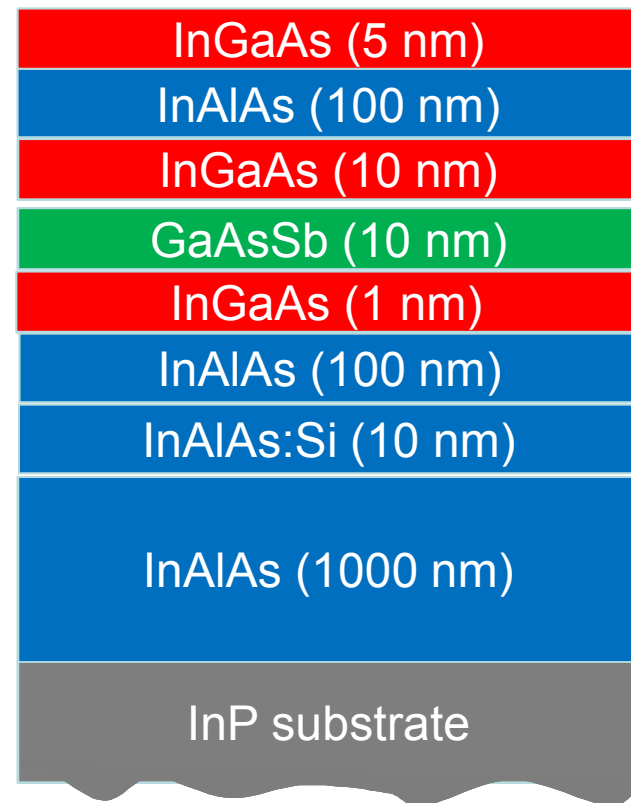
# Structure and analysis details

## 1) MQW with cap



- XRD
  - Symmetric (004)  $\Omega/2\Theta$  scans
  - Interpreted with dynamic simulations
- Room temperature PL
  - Taken with FTIR system using 1550 nm excitation and InSb detector

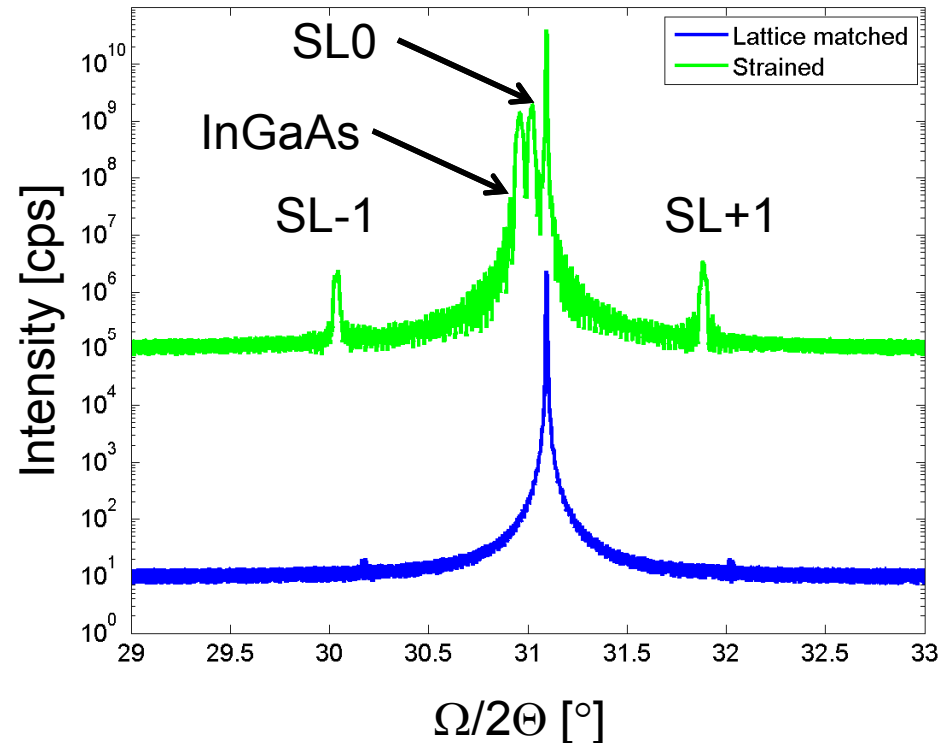
## 2) Quantum well channel



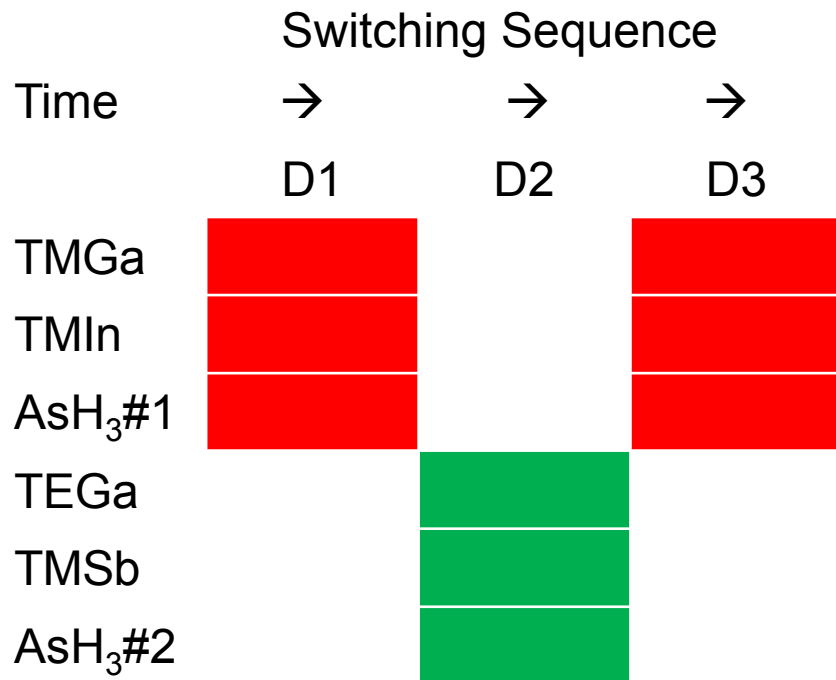
- XTEM
  - high resolution

# Simulations of ideal MQW

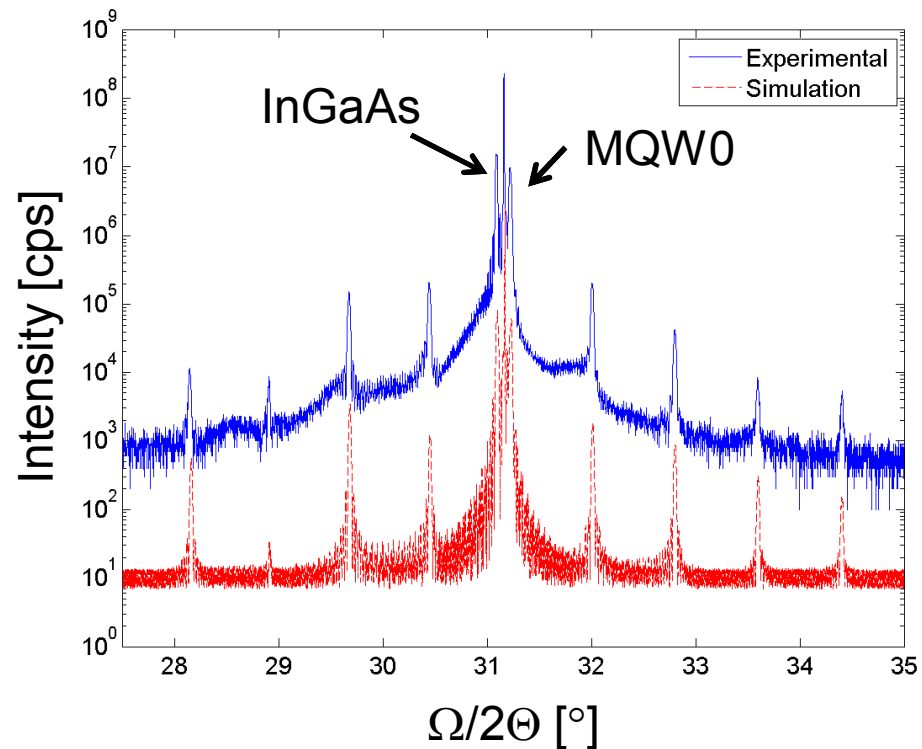
- Lattice matched InGaAs/GaAsSb have few features
  - Weak +/- 1 satellites are associated with small strain ( $< 50$  ppm)
- Strain introduces expected features
  - +3000 ppm for InGaAs
  - +1000 ppm for GaAsSb



# Initial MQWs - nonidealities



- MQW is in tension
- Graded GaAsSb layer varies from GaAs to GaAs<sub>0.44</sub>Sb<sub>0.56</sub>
- Additional non-monotonic intensity envelope observed at higher angle



Period of simulated MQW

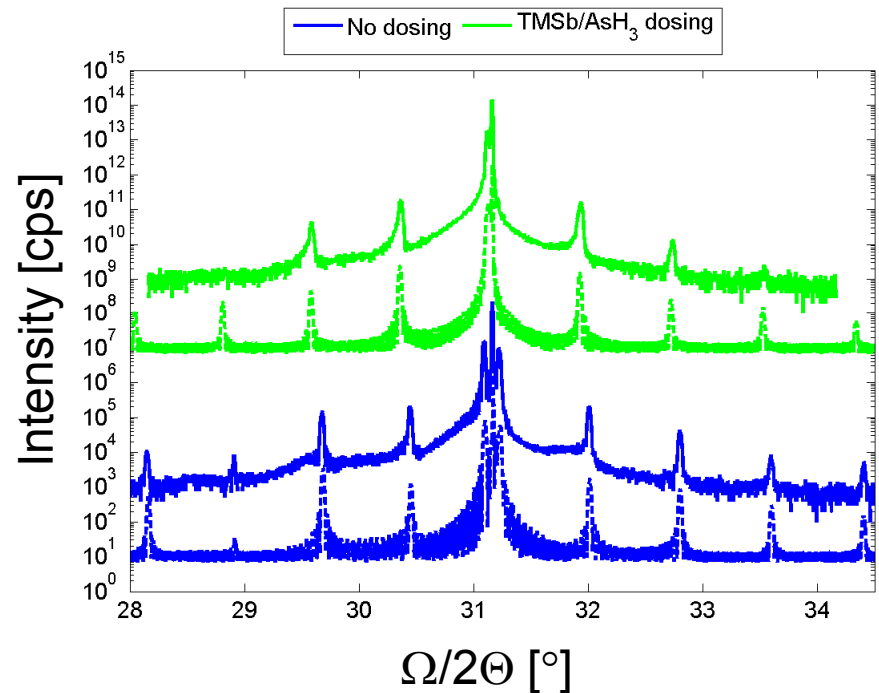
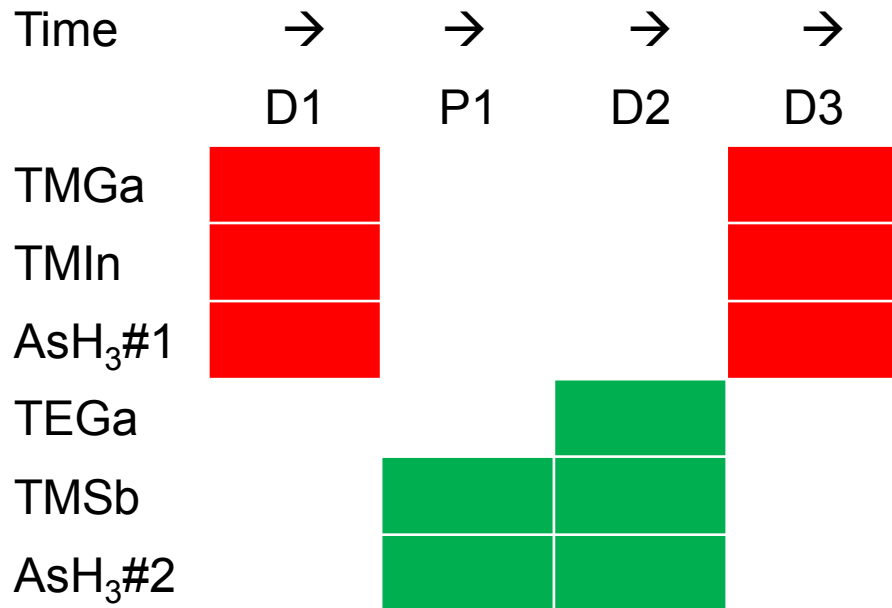




# TMSb/AsH<sub>3</sub> dosing of InGaAs surface

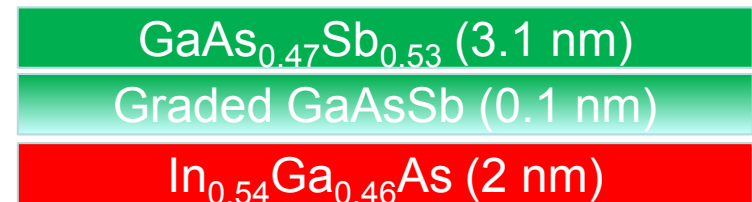


## Switching Sequence



- Investigated P1 from 3 to 24 s
  - Found 12 s worked well
- Dosing reduces, but doesn't eliminate strain

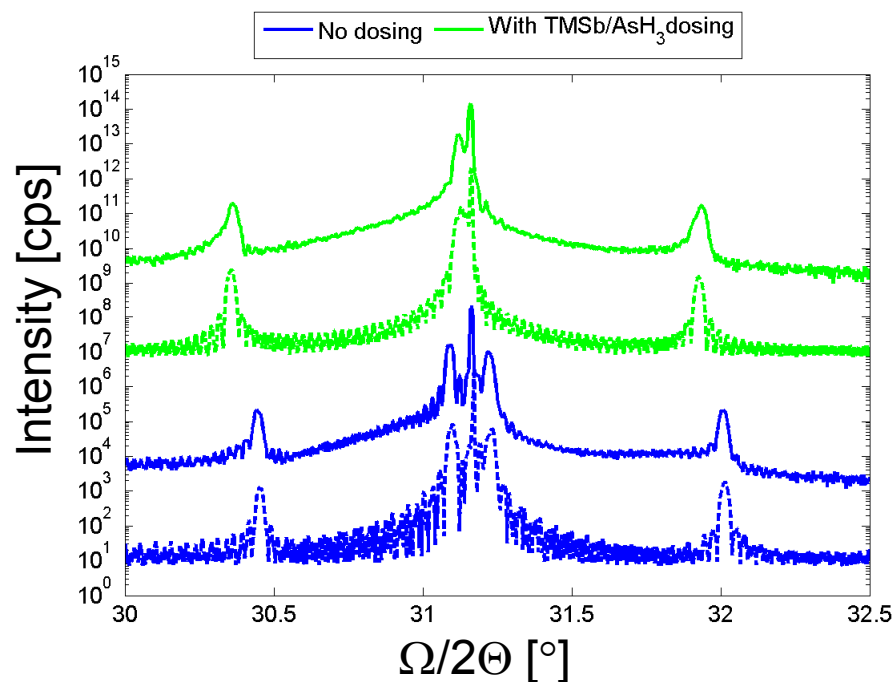
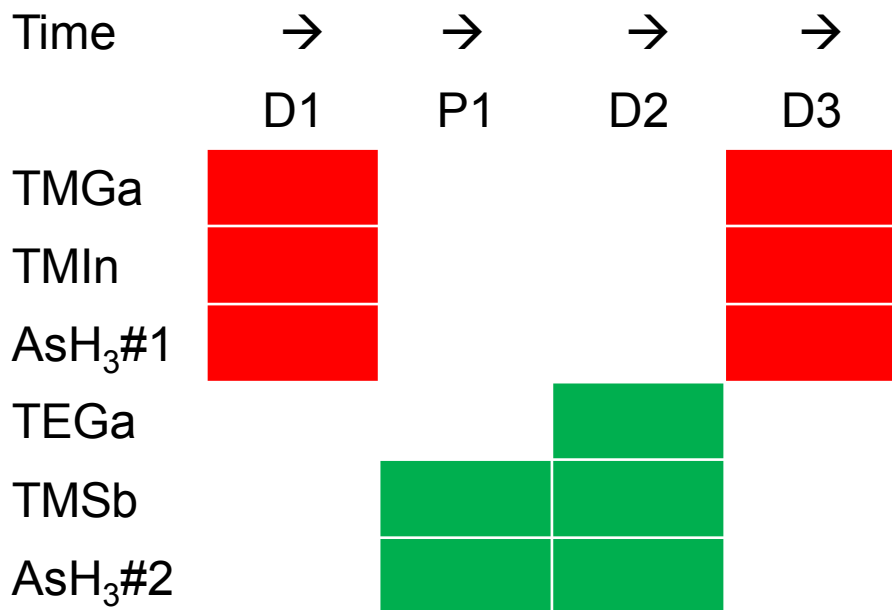
## Period of simulated MQW



# TMSb/AsH<sub>3</sub> dosing of InGaAs surface

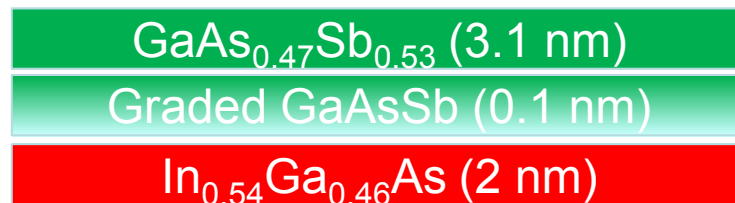


## Switching Sequence



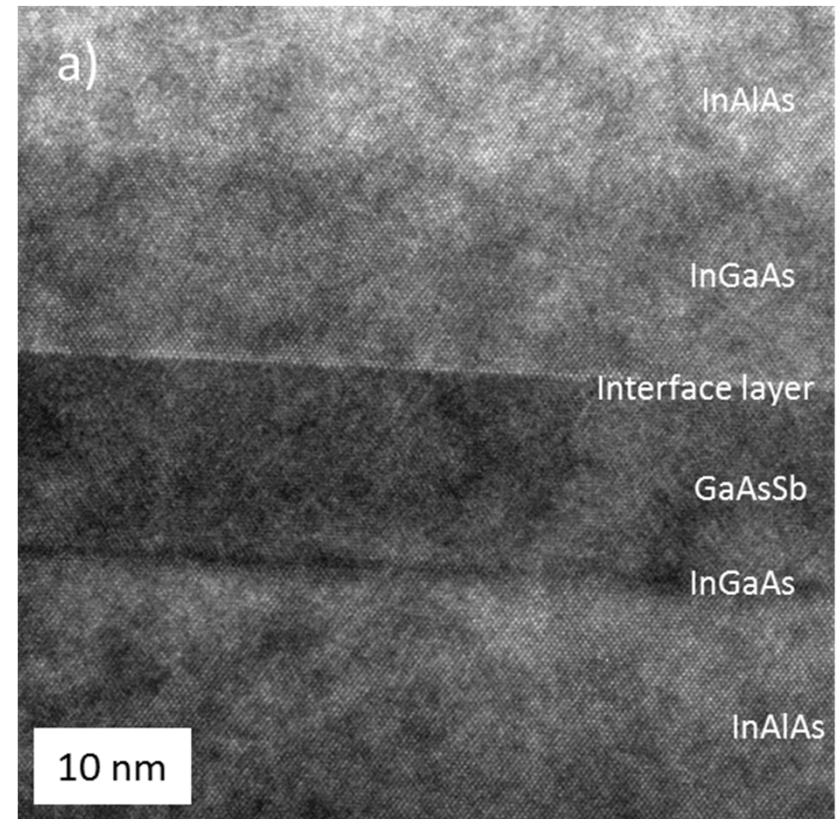
- Investigated P1 from 3 to 24 sec
  - Found 12 s worked well
- Dosing reduces, but doesn't eliminate strain

Period of simulated MQW



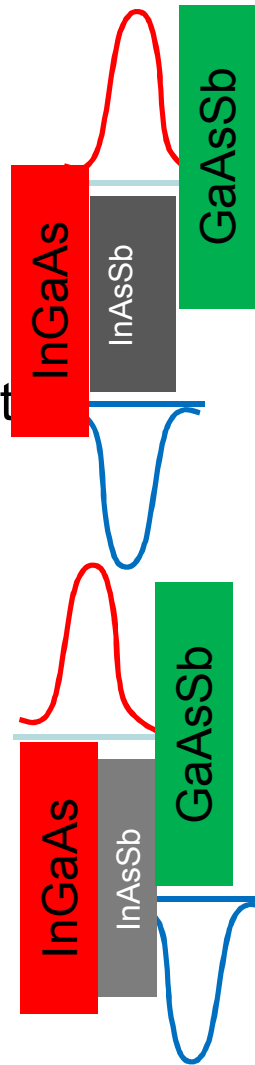
# XTEM on QW channel structure

- Using optimized TMSb/AsH<sub>3</sub> purge
- High resolution images are sensitive to difference in electron phase
  - Phase differences can be caused by strain
- Concluded that tensile layer is on opposite side of GaAsSb
  - GaAs-rich interface

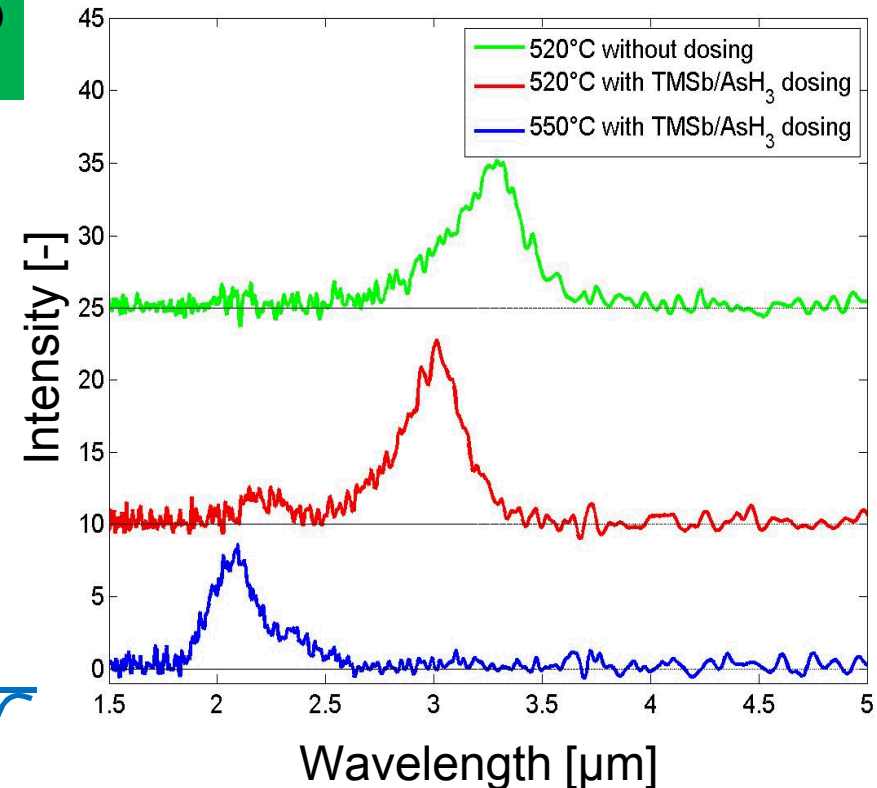


# PL on MQWs

- Samples grown at 520°C show 3+  $\mu\text{m}$  emission
  - Wavelength inconsistent with thicknesses and compositions
  - Attribute to interfacial InAsSb dominating recombination
- Samples grown at 550°C emit at 2  $\mu\text{m}$ 
  - Shoulder at 2.4  $\mu\text{m}$
- Found 570°C was too warm for bulk GaAsSb growth

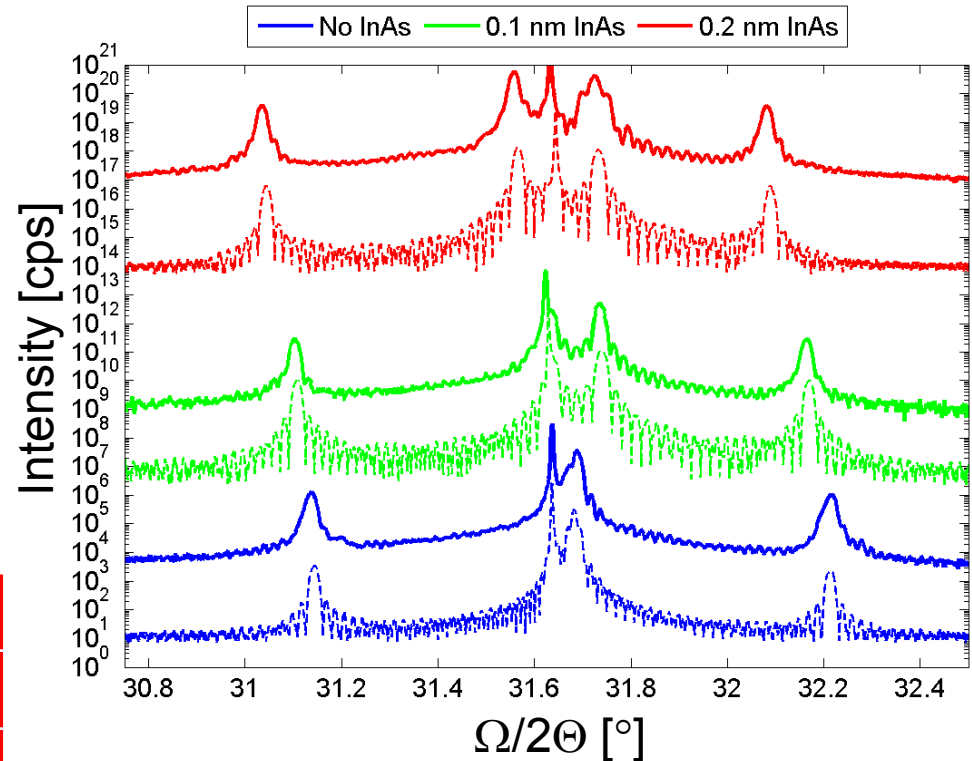
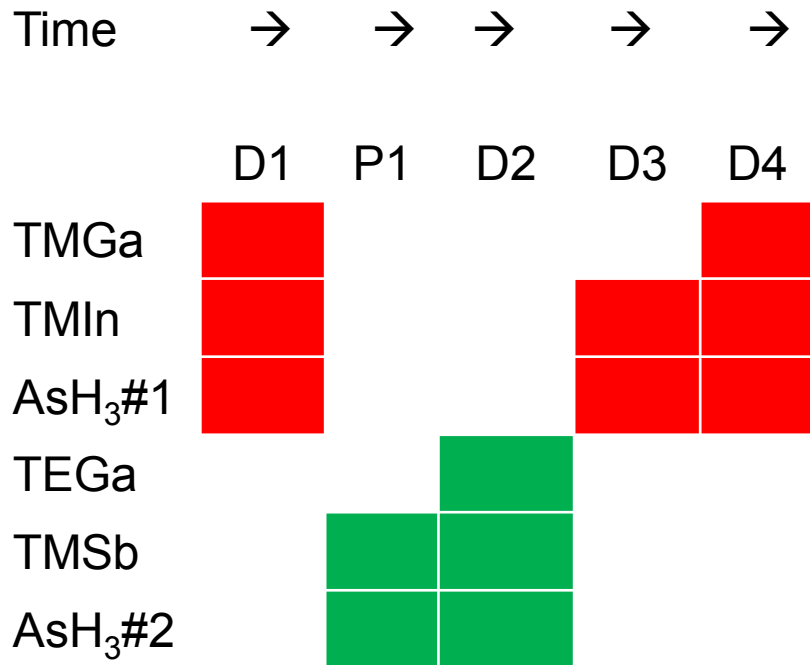


3.1 nm  $\text{GaAs}_{0.47}\text{Sb}_{0.53}$  / 2 nm  $\text{In}_{0.54}\text{Ga}_{0.46}\text{As}$



# Further strain reduction – InAs layers

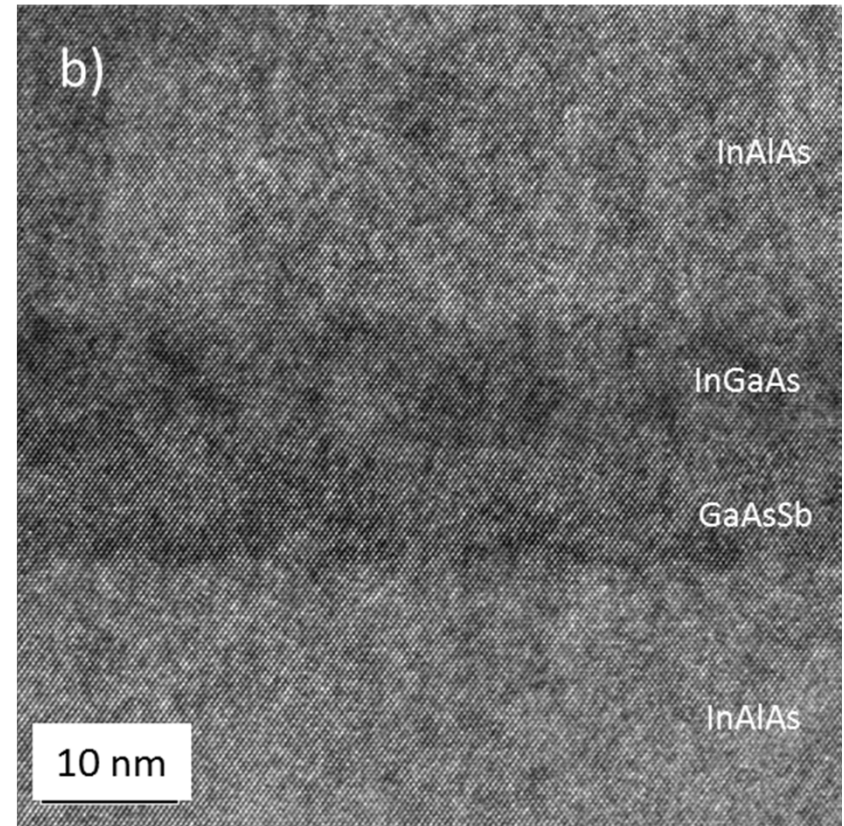
- Grow thin layers of InAs on top of GaAsSb
  - 0.1 nm optimal





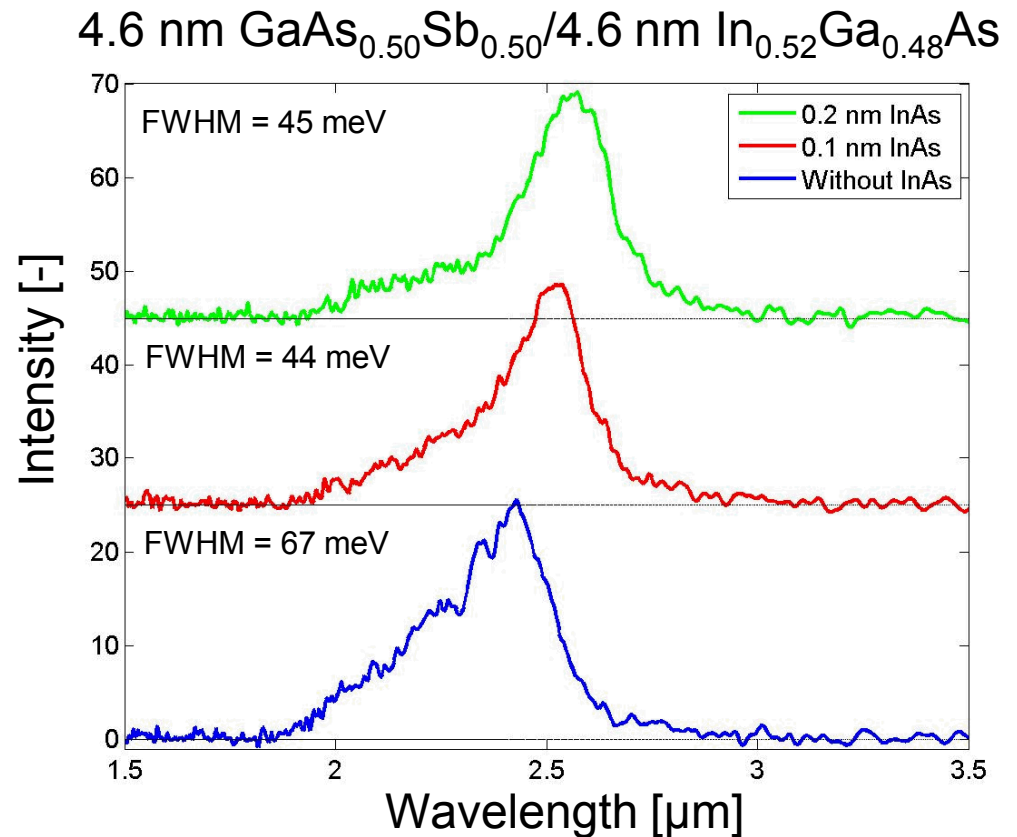
# XTEM on improved structure

- InAs growth eliminates transition at GaAsSb to InGaAs interface
  - Small contrast between InGaAs and GaAsSb consistent with matched compositions
  - Contrast consistent with mass density



# PL on InAs interface layer series

- Slight red-shift as thickness of InGaAs is increased
- Linewidth decreases with addition of InAs



# Conclusions

- InGaAs/GaAsSb represent a challenging system to achieve ideal heterostructures
- Sb-incorporation limits abruptness going from InGaAs to GaAsSb
- PL dominated by interfaces
  - Low band gap InAsSb dominates PL at low growth temperatures
  - Higher growth temperatures improves optical properties
- In-incorporation limits abruptness going from GaAsSb to InGaAs
- PL characteristics are improved by optimized interfaces