

## Ab initio study of zirconium(IV) chloride

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### INTRODUCTION

Zirconium halides are of crucial importance in numerous research fields and industrial applications such as unconventional catalysis [1], refining of Zr-containing ores by Kroll reduction [2], chemical vapor deposition (CVD) [3], or nuclear engineering [4-6]. For example, chlorination has been proposed for large-scale separation and selective recovery of Zr, as  $ZrCl_4$ , from U-Zr alloys or used nuclear fuel cladding [4, 5]. In addition,  $ZrX_4$  ( $X=Br, Cl$ ) is utilized in the preparation of CVD zirconium carbide layers of tristructural-isotropic (TRISO) nuclear fuel micro-particles [6]. Although an accurate knowledge of the properties of zirconium halides is key to optimizing process conditions for the aforementioned applications, recent and accurate thermomechanical information for Zr halides remains scarce [3]. In particular, for crystalline  $ZrCl_4$  the low-temperature calorimetric measurements by Todd [7] and Efimov et al. [8], in the temperature ranges 52 – 296 K and 9 – 315 K, are still used as references [3]. In addition, no comprehensive computational studies of the thermomechanical properties of  $ZrCl_4(cr)$  have been reported, to the best of our knowledge. Here, we report computational studies of the structural, lattice dynamics, and thermomechanical properties of bulk monoclinic  $ZrCl_4$  using the zero-damping dispersion-corrected density functional theory [DFT-D3(zero)], the quasi-harmonic approximation (QHA).

Total-energy calculations were carried out using Grimme's dispersion-corrected DFT (DFT-D3)[9], as implemented in the Vienna *ab initio* simulation package (VASP) [10]. The exchange-correlation energy was computed using the generalized gradient approximation (GGA), with the parameterization of Perdew, Burke, and Ernzerhof (PBE) [11]. Previous first-principles studies demonstrated that standard functionals, such as PBE, correctly reproduce the structure-properties relationship of bulk zirconium and Zr alloys, as well as transition-metal chloride compounds [12-15].

The projector augmented wave (PAW) method [16,17] was utilized to model the interaction between valence electrons and ionic cores. PAW pseudopotentials were used to represent the remaining core electrons together with the nuclei. The Kohn-Sham (KS)

equations were solved using the blocked Davidson iterative matrix diagonalization scheme [18]. A plane-wave cutoff energy of 500 eV was chosen for the electronic wavefunctions, ensuring total-energy convergence to within 1 meV/atom. Partial occupancies of the wavefunctions were controlled using Gaussian smearing, with a Gaussian width of 0.1 eV. Simultaneous ionic and cell relaxations of  $ZrCl_4$  were conducted, without symmetry constraints applied. The Hellmann-Feynman forces acting on atoms were calculated with a convergence tolerance set to 0.01 eV/Å. The Brillouin zone (BZ) was sampled using the Monkhorst-Pack  $k$ -point scheme [19] with a  $k$ -point mesh of  $5 \times 5 \times 5$ .

### RESULTS

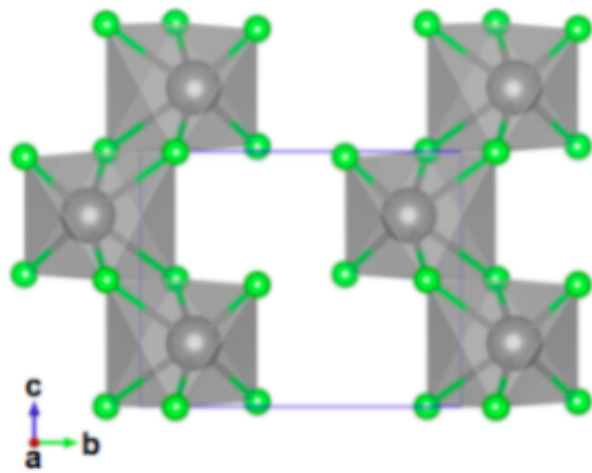
Initial structural optimization calculations with standard DFT-PBE for the bulk monoclinic  $ZrCl_4$  unit cell (space group  $P2_1/c$ ; IT No. 13;  $Z = 2$ ) yielded lattice parameters of  $a = 6.93$ ,  $b = 8.16$  and  $c = 6.35$  Å ( $\alpha = \gamma = 90^\circ$ ,  $\beta = 107.6^\circ$ ;  $V = 342.0$  Å<sup>3</sup>), considerably larger than X-ray diffraction (XRD) measurements [20]. This stems from the lack of a correct description of cohesive van der Waals interactions resulting from dynamical correlations between fluctuating charge distributions in standard DFT. Therefore, Grimme's dispersion-corrected DFT (DFT-D3) was utilized in this study. Table 1 summarizes the calculated and experimental structural parameters.

**Table 1** The lattice parameters of bulk monoclinic  $ZrCl_4$  calculated in the zero-temperature limit.

$ZrCl_4$ ( $P2_1/c$ )	DFT-PBE (this work)	XRD [20]	DFT-D3 (this work)
a (Å)	6.39	6.361 ± 0.004	6.38
b (Å)	8.16	7.407 ± 0.004	7.49
c (Å)	6.35	6.256 ± 0.004	6.25

$\beta$ ( $^\circ$ )	107.6	109.3 $\pm$ 0.04	107.6
Volume ( $\text{\AA}^3$ )	342	278.2	282.9

The lattice parameters of bulk monoclinic  $\text{ZrCl}_4$  calculated in the zero-temperature limit are in excellent agreement with experimental values from XRD (X-ray diffraction) measurements [20] as shown in Table 1. The optimized  $\text{ZrCl}_4$  unit-cell, featuring chains of edge-sharing coordination octahedra along the [001] direction, is depicted in Figure 1.



**Figure 1** Crystal structure of bulk  $\text{ZrCl}_4$  (space group P2/c; IT No. 13;  $Z = 2$ ) optimized with DFT-D3(zer). The unit cell is represented by solid blue lines. Color legend: Cl, green; Zr, grey.

The computed bond distances between Zr centers (2e Wyckoff positions) and apical and bridge Cl ligands (4g Wyckoff positions) are  $d_{\text{Zr}-\text{Cl}_{\text{apical}}} = 2.33 \text{ \AA}$  and  $d_{\text{Zr}-\text{Cl}_{\text{bridge}}} = 2.50, 2.69 \text{ \AA}$ . These bond distances compare well with the XRD values of  $d_{\text{Zr}-\text{Cl}_{\text{apical}}} = 2.307 \text{ \AA}$  and  $d_{\text{Zr}-\text{Cl}_{\text{bridge}}} = 2.497, 2.656 \text{ \AA}$  [20]. The calculated (measured) Zr–Zr distance is  $3.99 \text{ \AA}$  ( $3.962 \text{ \AA}$ ), and the bond angles  $\angle \text{Cl}_{\text{apical}}\text{ZrCl}_{\text{apical}}$  and  $\angle \text{Cl}_{\text{bridge}}\text{ZrCl}_{\text{bridge}}$  are  $100.9$  and  $79.6^\circ$  ( $100.66$  and  $79.54^\circ$ ), respectively.

Phonon analysis was conducted using the finite-displacement method near equilibrium volume within the QHA in order to derive thermal properties of bulk  $\text{ZrCl}_4$ . A temperature effect was added to the calculated total energy  $U(V)$  of the system through the phonon contribution. While the melting point of  $\text{ZrCl}_4$  was reported as  $710 \pm 1 \text{ K}$ , a monoclinic to cubic phase

transformation around  $\approx 538 \text{ K}$  was discussed in previous studies [3]. Therefore, the present calculations for monoclinic  $\text{ZrCl}_4(\text{cr})$  are limited to temperatures below  $\approx 550 \text{ K}$ . The computed thermal evolutions of  $K_0$  and  $K_0'$  using the Vinet equation of state [22]. The bulk modulus and its pressure derivative are predicted to vary from  $K_0 = 8.7$  to  $7.0 \text{ GPa}$  and from  $K_0' = 10.4$  to  $8.9 \text{ GPa}$  between 0 and  $550 \text{ K}$ . Interestingly, both  $K_0$  and  $K_0'$  do not decrease monotonically with temperature, but exhibit instead maxima in the vicinity of  $\approx 20$  and  $80 \text{ K}$ , respectively.

The isobaric molar heat capacity ( $C_p$ ) and entropy ( $S$ ) are calculated and compared with the available experimental data [27]. It is found that our calculated entropy, isochoric molar heat capacity, and isobaric molar heat capacity at standard pressure ( $P = 1 \text{ bar}$ ) for bulk  $\text{ZrCl}_4$  are in excellent agreement with low-temperature heat capacity measurements by Todd [7] and Efimov et al. [8]. The standard values calculated at  $T = 298.15 \text{ K}$  in this study are  $C_p^0 = 107.3 \text{ J/mol K}^{-1}$  and  $S_p^0 = 162.1 \text{ J mol}^{-1} \text{ K}^{-1}$ .

In Summary, DFT-D3(zer) calculations were conducted to investigate the lattice dynamics and thermo-mechanical properties of bulk monoclinic  $\text{ZrCl}_4(\text{cr})$ . The isobaric molar heat capacity derived from phonon calculations within the quasi-harmonic approximation is in fair agreement with existing calorimetric data. New heat-capacity measurements for high-purity  $\text{ZrCl}_4(\text{cr})$  are desirable, especially above room temperature where calorimetric data are scarce.

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## ACKNOWLEDGEMENT

This research was performed using funding received from the DOE Office of Nuclear Energy's Nuclear Energy University Program. Sandia National Laboratories is a multi-mission laboratory managed and operated by National Technology and Engineering Solutions of Sandia, LLC., a wholly owned subsidiary of Honeywell International, Inc., for the U.S. Department of Energy's National Nuclear Security.