

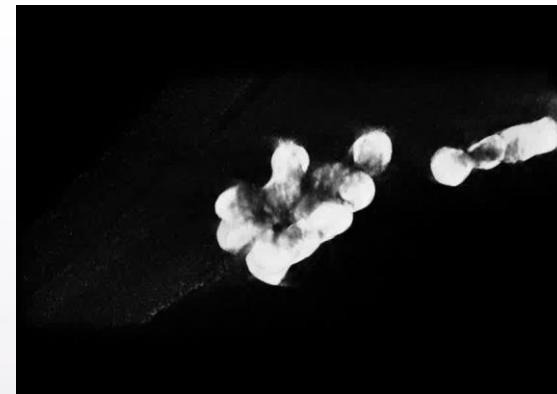
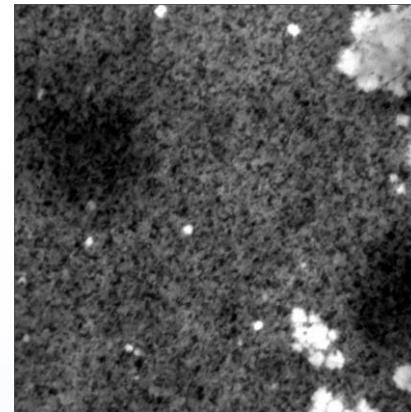
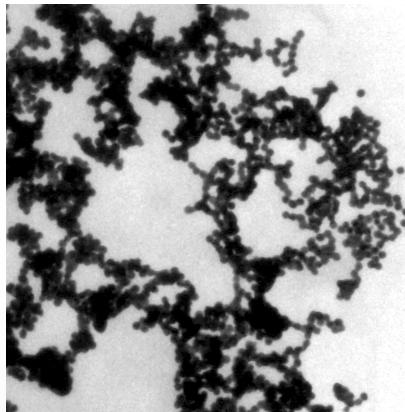
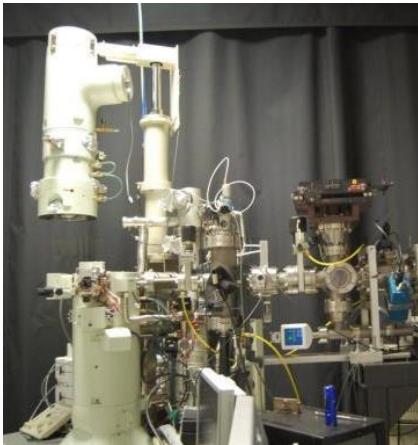
Exploring Materials in Extreme Environments with Nanometer Resolution

SAND2017-13003PE

K. Hattar

Sandia National Laboratories

November 14, 2017



In situ TEM microscopy has recently undergone significant growth providing capabilities to investigate the structural evolution that occurs due to various extreme environments and combinations thereof

Collaborators:

- IBL: **D.C. Bufford, D. Buller, C. Chisholm, B.G. Clark, B.L. Doyle, S. H. Pratt, & M.T. Marshall**
- Sandia: **B. Boyce, T.J. Boyle, P.J. Cappillino, J.A. Scott, B.W. Jacobs, M.A. Hekmaty, D.B. Robinson, E. Carnes, J. Brinker, D. Sasaki, J.A. Sharon, T. Nenoff, W.M. Mook**
- External: **A. Minor, L.R. Parent, I. Arslan, H. Bei, E.P. George, P. Hosemann, D. Gross, J. Kacher, & I.M. Robertson**

This work was supported by the US Department of Energy, Office of Basic Energy Sciences.

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Potential Evolution of System Design

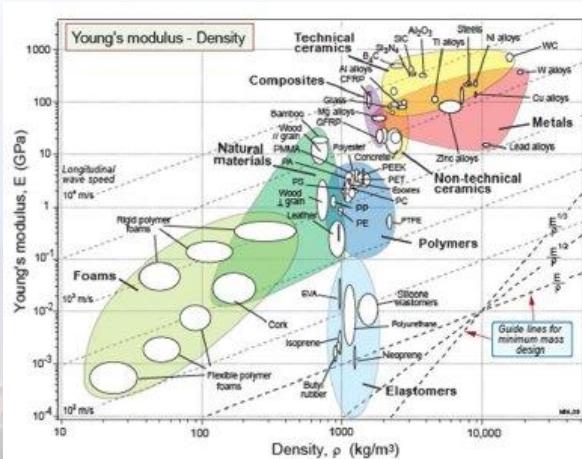
Use the Nearest Stone



to



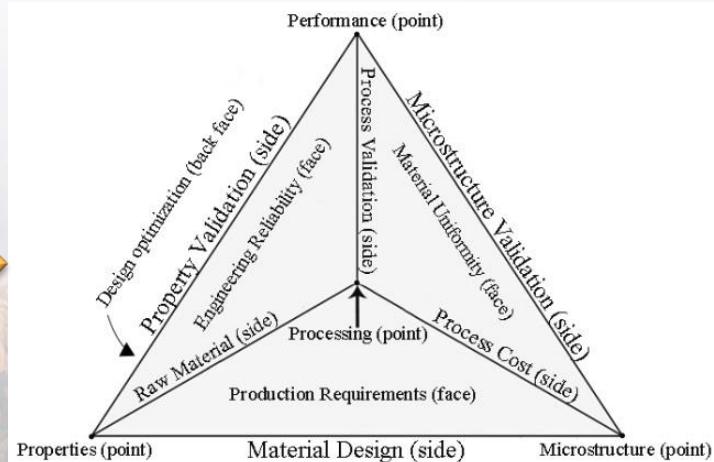
- Radar charts and Ashby plots of current material
- Accelerated and field testing
- **Scientist create a new materials. Engineers find an application.**



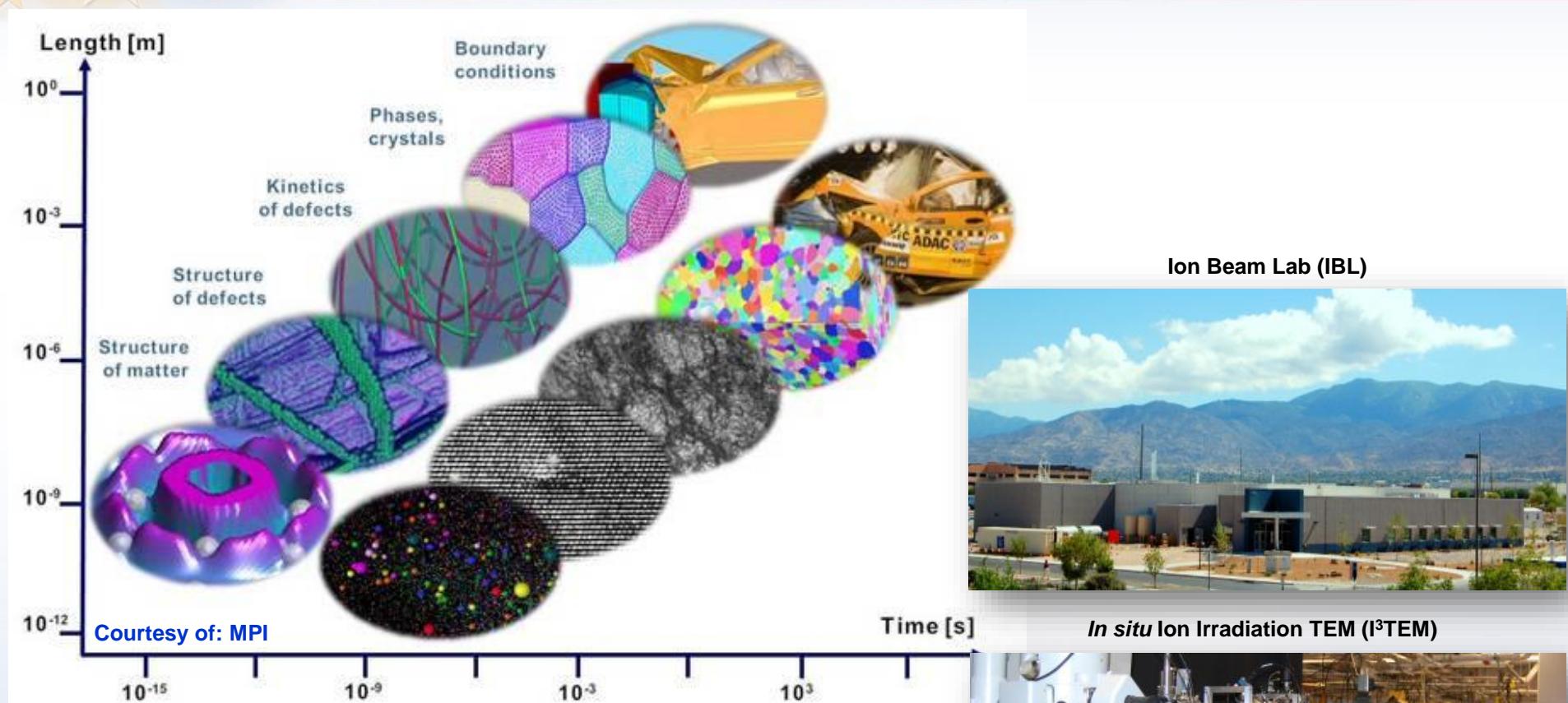
Materials by Design

- Physics-based approach
- Requires multiscale modeling
- **Engineers require given properties, Scientists tailor the chemistry and microstructure to achieve it.**

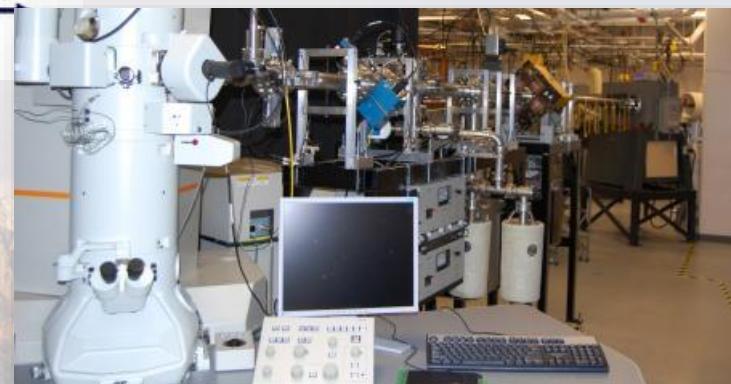
Great vision! We are making strides, but we are not there yet



Investigating the **nm** Scale to Understand the **km** Scale Response of Materials in the Extremes



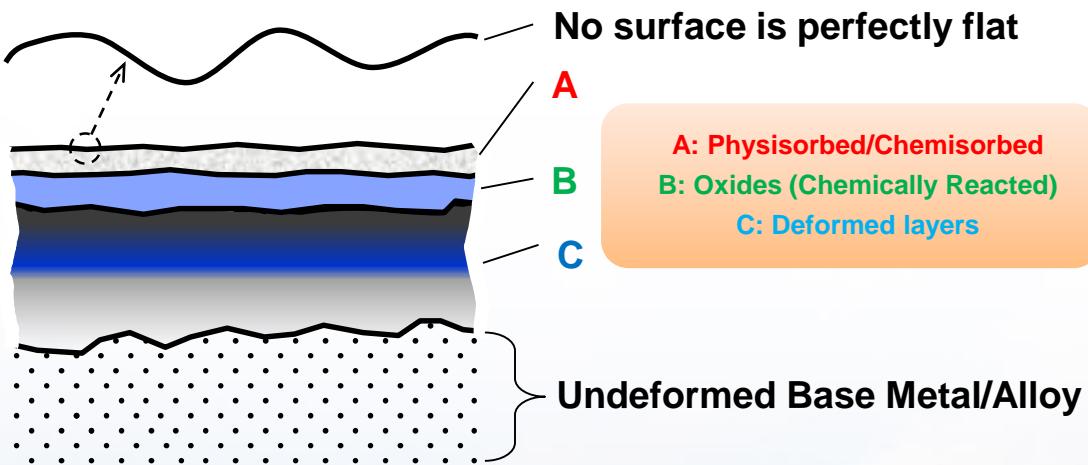
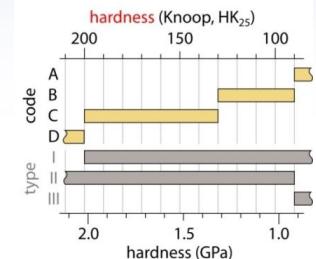
To develop predictive physics-based models, a fundamental understanding of the structure of matter, defects, and the kinetics of structural evolution in the environments of interest are needed



Nature of Metallic Surfaces

Electronics Applications (connectors, switches and relays):

- Gold desired because it is chemically Inert and has low electrical resistance
- Pure gold is soft and has unacceptable amount of friction and wear
- Gold is typically hardened by alloying with small amounts of Ni or Co

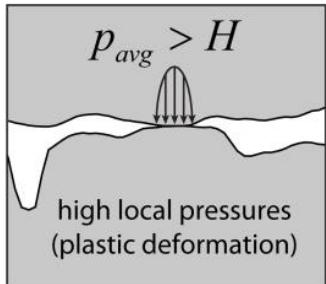


From ASTM B488-11 / MIL-DTL-45204D:

type	suggested applications (ASTM)
> 99.7% Au	I general-purpose, high-reliability electrical contacts
(hardest)	
> 99.0% Au	II general-purpose, wear resistance; low temperature only
(softest)	
> 99.9% Au	III soldering; limits impact of oxidation of codeposited material
	III A semiconductor components, nuclear eng., high temperature

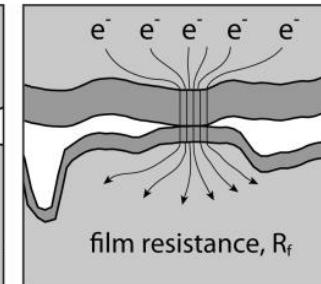
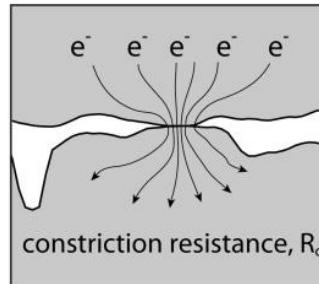
Asperity Contacts, Constriction, Asperity Contacts and Surface Films

areal sum of asperity contacts and surface films define electrical contact resistance



... for metal contacts the real area is a function of hardness and contact force (Bowden & Tabor, 1939):

$$A_r \approx \frac{F_n}{H}$$



... ECR is a function of the constriction and film resistances:

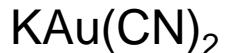
$$ECR = \sum_i (R_{c,i} + R_{f,i})$$

Archard, JAP (1953) 24:981
 R. Holm, Electrical Contacts Handbook (1958)
 Greenwood & Williamson, P. Proc. Royal Society (1966)
 T.W. Scharf & S.V. Prasad, JMS (2013) 48:511-531

Can We Get Away from Electrochemical Hard Au?

Electroplating:

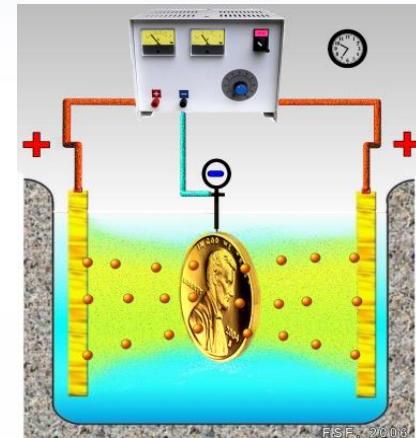
Hard Gold is typically plated from a cyanide precursor



Can contain bath modifiers of As, Pb, Cd or Tl

Small amounts hardening agent such as Ni or Co

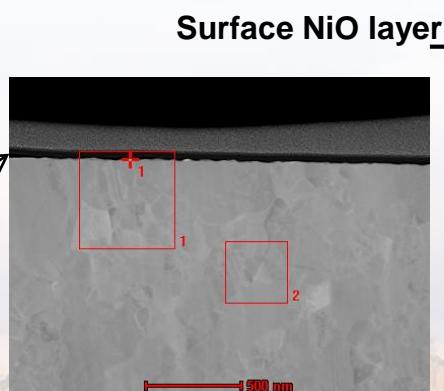
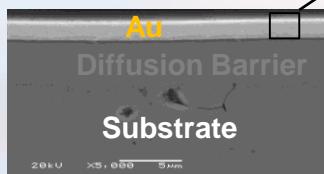
Nickel barrier layer applied first



Diffusion and Surface Oxidation:

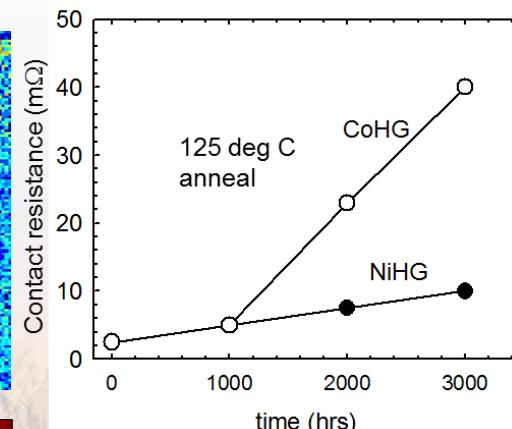
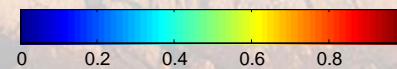
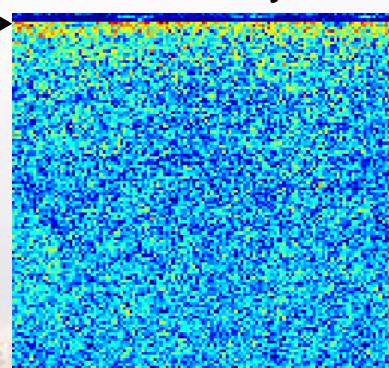
Diffusion and segregation of Ni or Co hardeners to the surface over time, ECR degradation

Electroplated Hard Au does not Age well and can be Toxic!



Cross-sectional TEM and Spectral Imaging

Relative Ni concentration
After 35+ years



Y. Okinaka and M. Hoshino, Gold Bulletin, 31(1), 3 (1998).

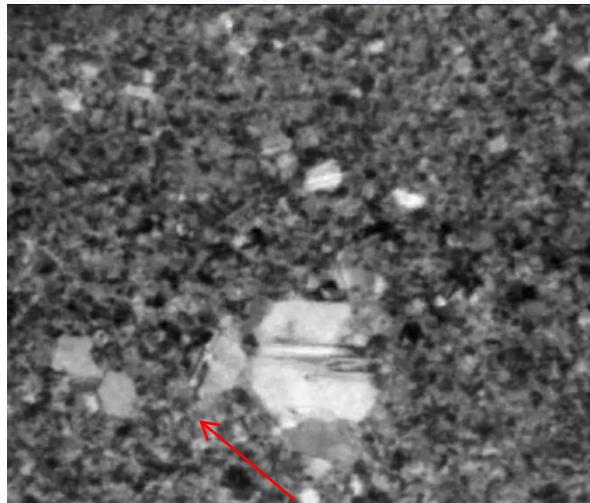


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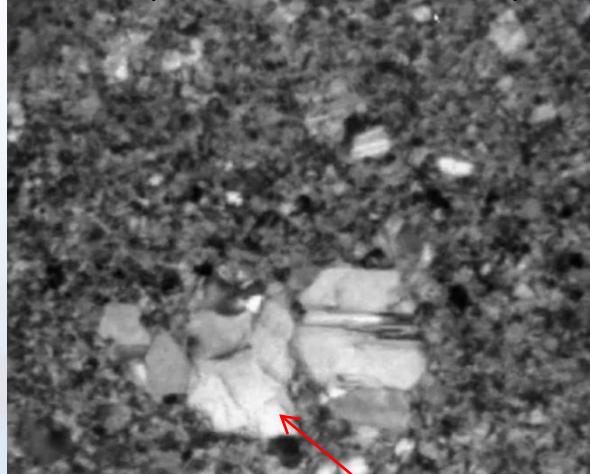
Exploring the Thermal Stability of PVD Hard Gold

Collaborators: J-E Mogonye, S.V. Prasad, J. Machael, R. Goeke, and P. Kotula

Before

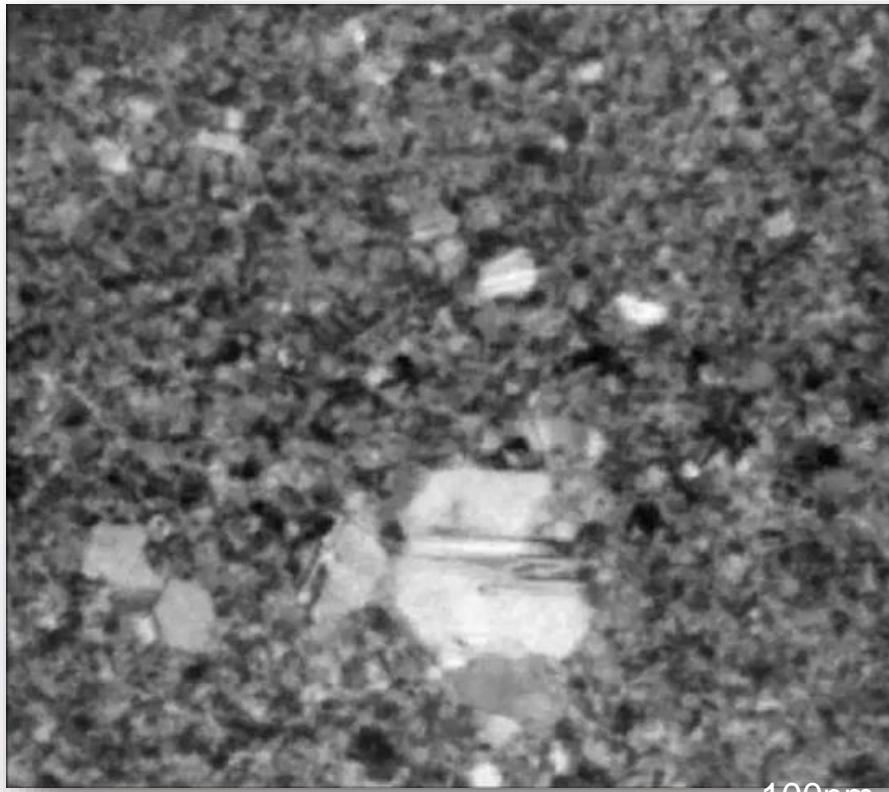


After (2 min @ 155° C)

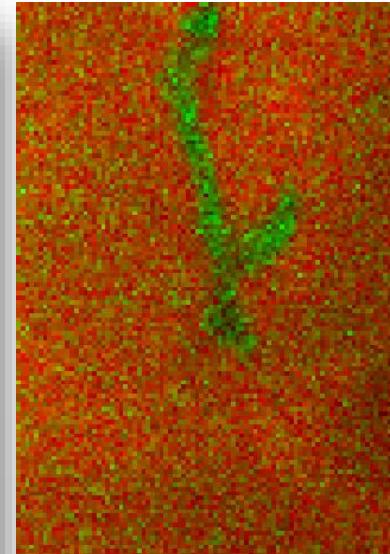


Abnormal grain growth dominates the thermal stability of Au+ZnO

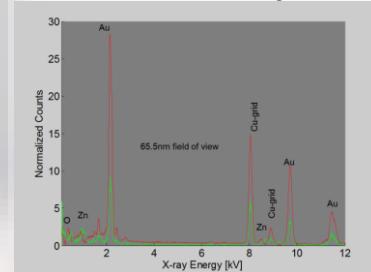
Au + ZnO 5%



Zn/Au ratio



65.5nm vertical length



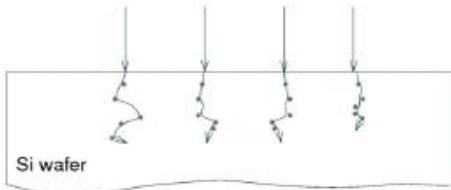
- Au+ZnO can be co-deposition by PVD (e-beam evaporation or sputtering)
- Au+ZnO shows good tribology properties
- PVD is environmentally friendly



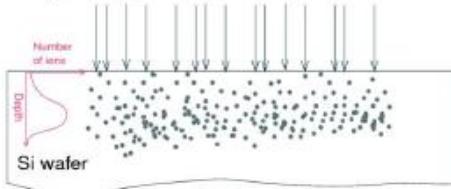
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Previous Studies Investigated Embedded Particles

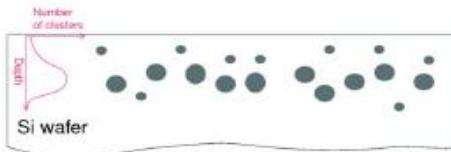
1. Ion implantation



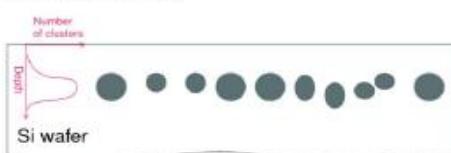
2. Implantation profiles



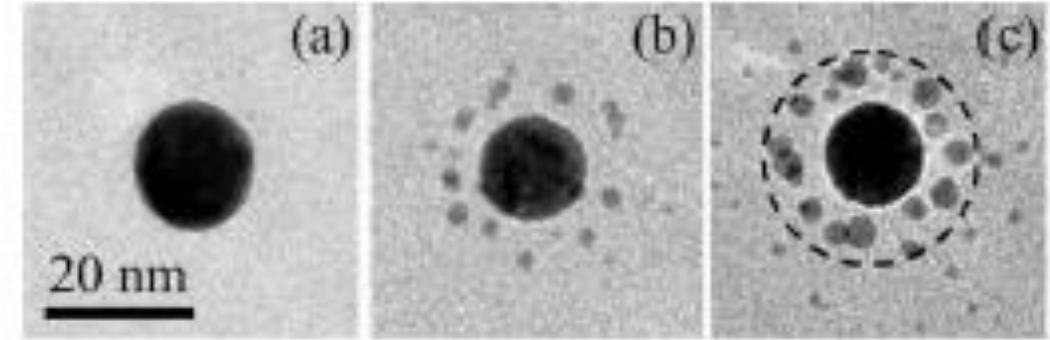
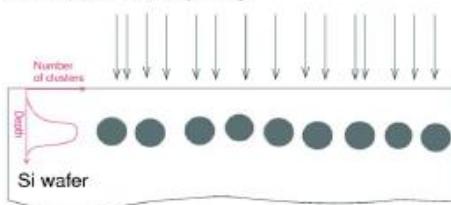
3. Nanocluster formation



4. Ostwald ripening



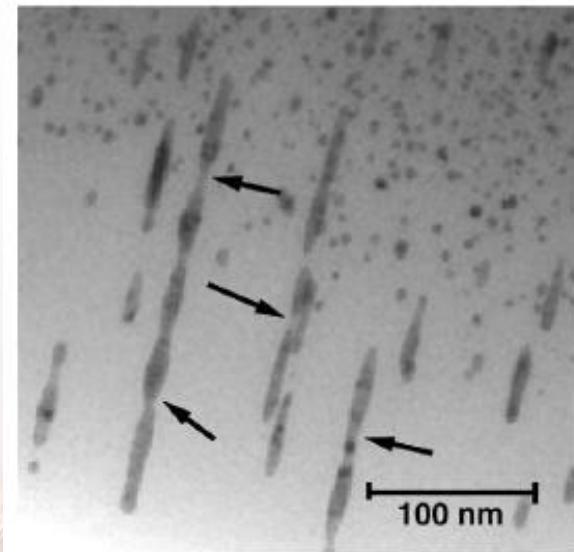
5. Inverse Ostwald ripening



G. Rizza et. al. *Phys. Rev. B* 76 (2007) 245414.

Au NCs in SiO_2 matrix irradiated at increasing fluences with 4 MeV Au ions at room temperature. Two generations of satellites observed.

Ion tracks can be formed in SiO_2 matrices and nanoparticle elongation and nanowire formation has been observed. Pt NP are shown, irradiated with 185 MeV Au ions. Arrows point to nanorod fragments.



M.C. Ridgway et. al. *NIM B* 267 (2009) 931.



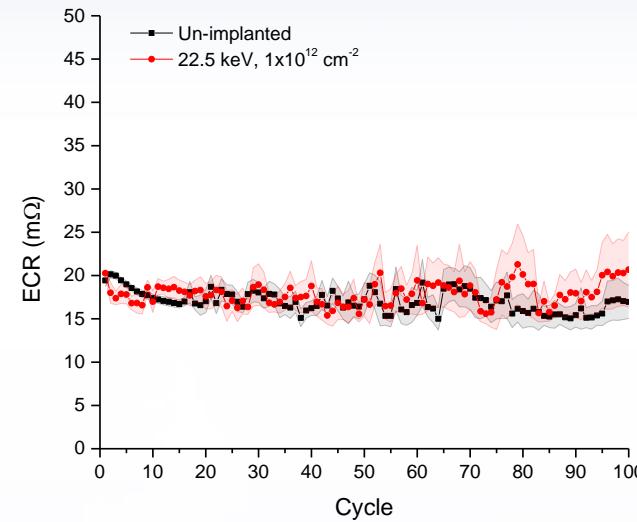
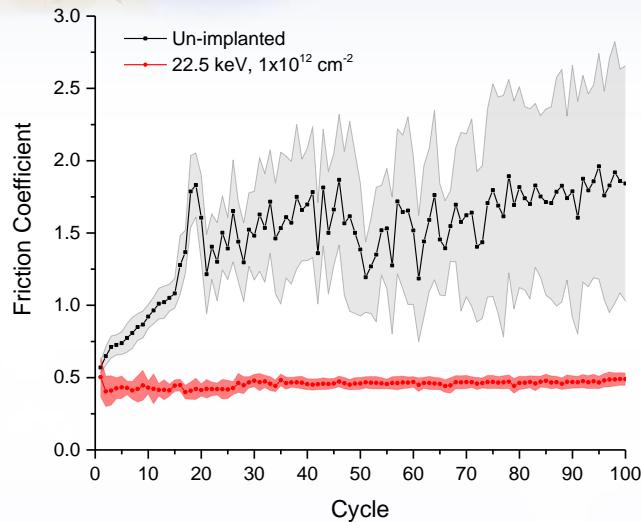
A.V. Krasheninnikov and K. Nordlund. *J. Appl. Phys.* 107 (2012) 071301.



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He Implantation Greatly Improves Wear Properties

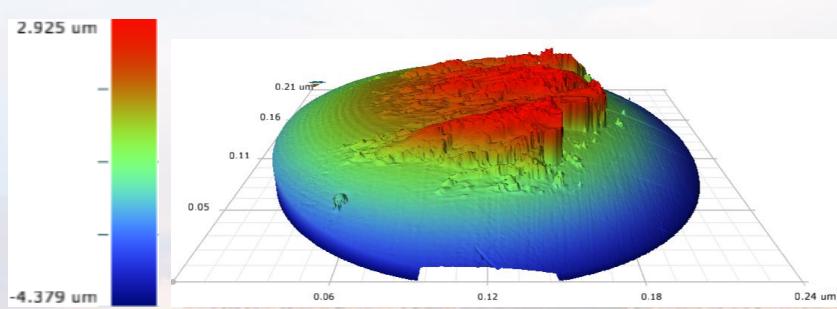
Collaborators: J-E Mogonye & S.V. Prasad



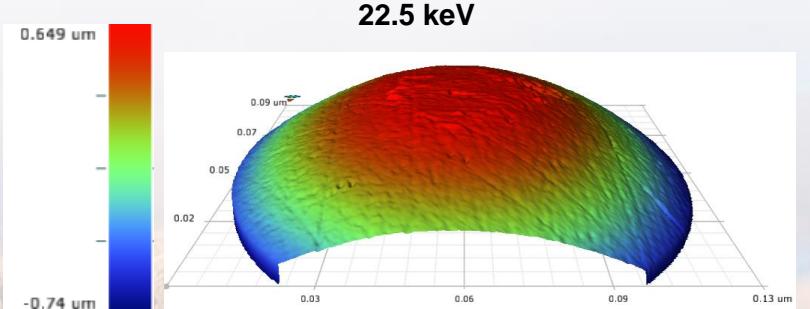
Friction is significantly reduced with ^3He implantation while maintaining ECR performance

Scanning white light interferometer topographical construction of riders after 100 cycles

Rider after 100 cycles against Un-implanted Au



Rider after 100 Cycles against Au implanted to $1\text{E}12 \text{ cm}^{-2}$ @ 22.5 keV



Wear is significantly reduced with minimal effect in ECR



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STEM Images of Sub-surfaces Provides Insight

Collaborators: P. Kotula, J-E Mogonye & S.V. Prasad

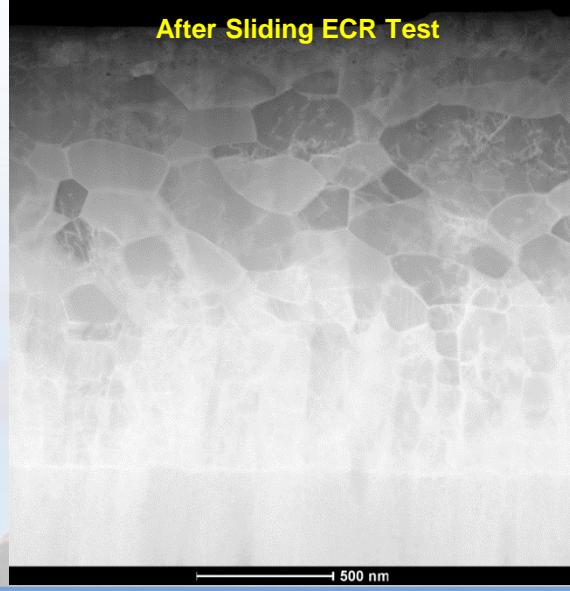
Before Sliding ECR Test



22.5 keV
 $1 \times 10^{12} \text{ cm}^{-2}$

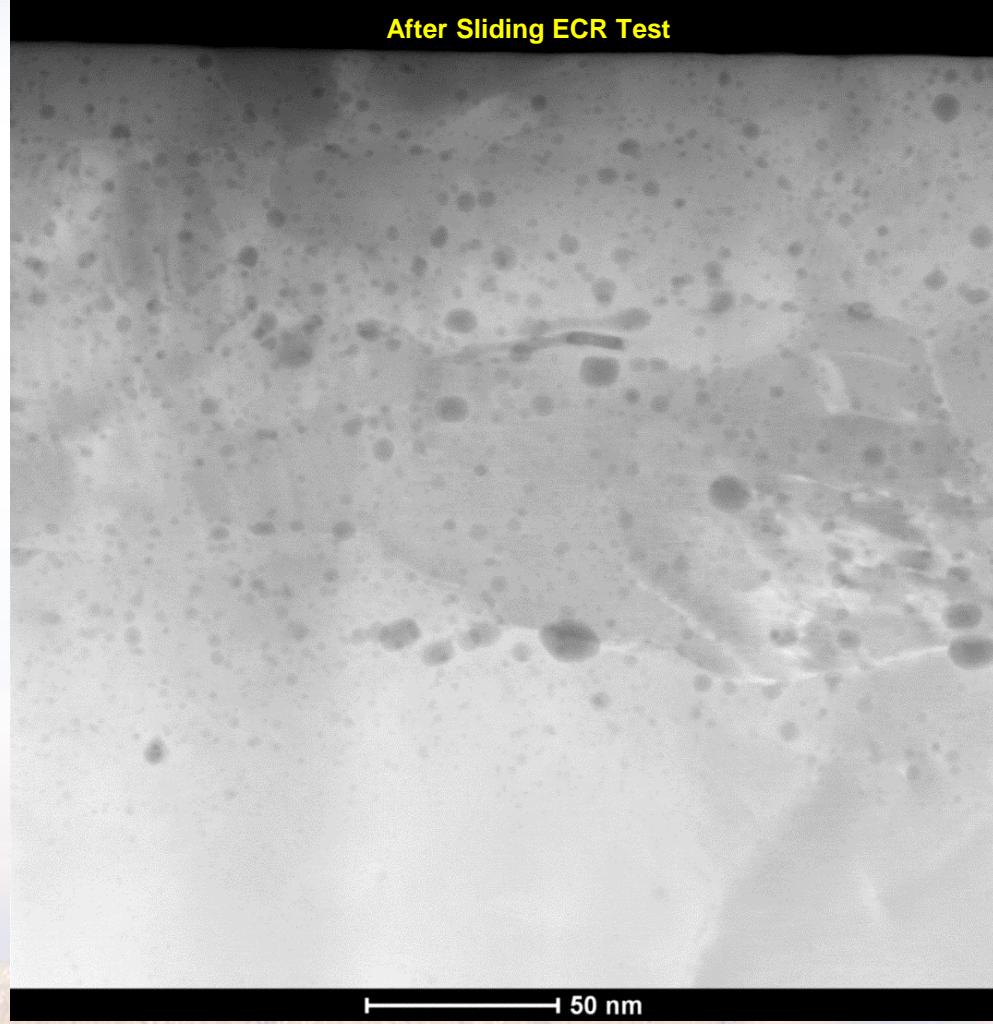
*Recrystallization
is observable
after 100 cycles*

After Sliding ECR Test



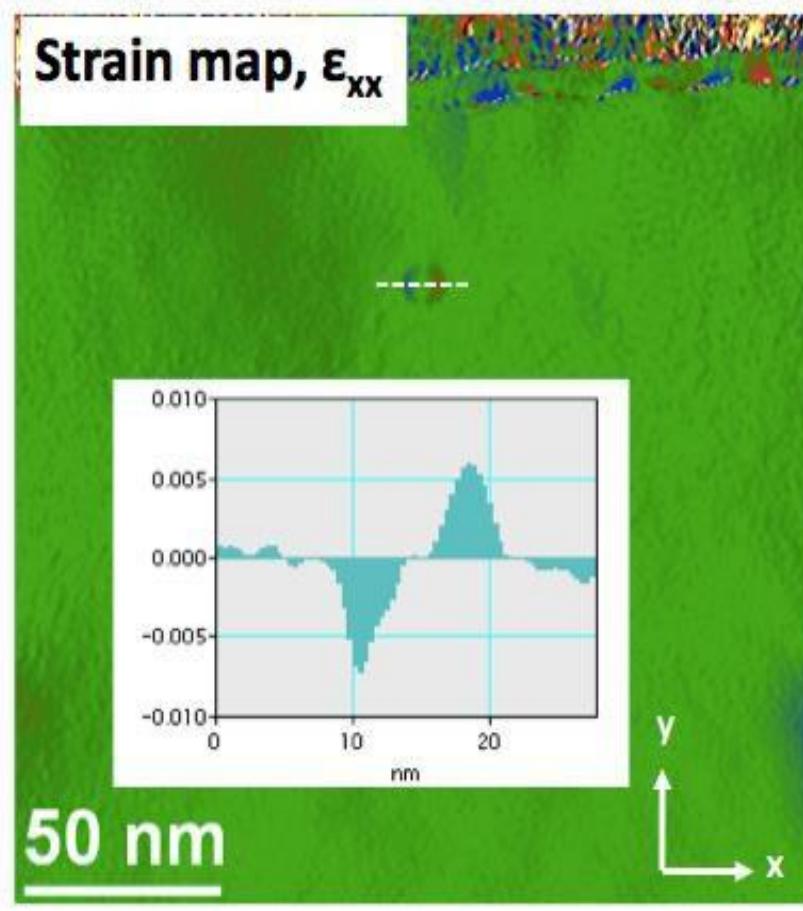
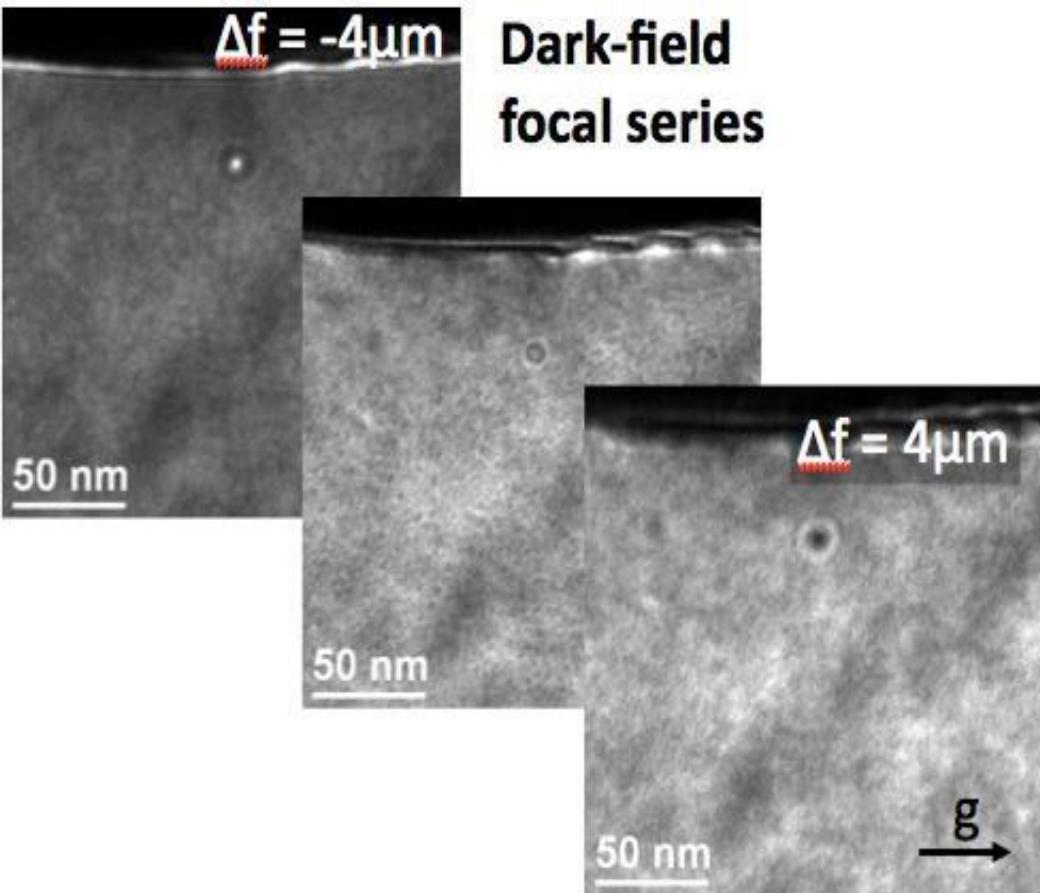
Au – Pt
Interface

After Sliding ECR Test



*An increase in both observable density and diameter of He bubbles,
suggests wear induced He coalescence from interstitial and previously
un-observable He*

Strain mapping at a He bubble in Au



In-line Holography Suggests that Significant Local Strain May Contribute to Enhanced GB Stability and Subsequent Wear Properties



Benefits & Limitations of *in situ* TEM

Benefits

1. Real-time nanoscale resolution observations of microstructural dynamics

Limitations

- 1. Predominantly limited to microstructural characterization**
 - Some work in thermal, optical, and mechanical properties
- 2. Limited to electron transparent films**
 - Can often prefer surface mechanisms to bulk mechanisms
 - Local stresses state in the sample is difficult to predict
- 3. Electron beam effects**
 - Radiolysis and Knock-on Damage
- 4. Vacuum conditions**
 - 10^{-7} Torr limits gas and liquid experiments feasibility
- 5. Local probing**
 - Portions of the world study is small

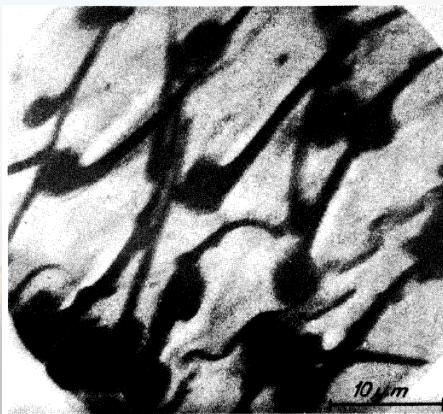


Fig. 6: Wing surface of the house fly.
(First internal photography, $U = 60$ kV, $M_u = 2200$)
(Driest, E., and Müller, H.O.: Z. Wiss. Mikroskopie 52, 53-57 (1935))

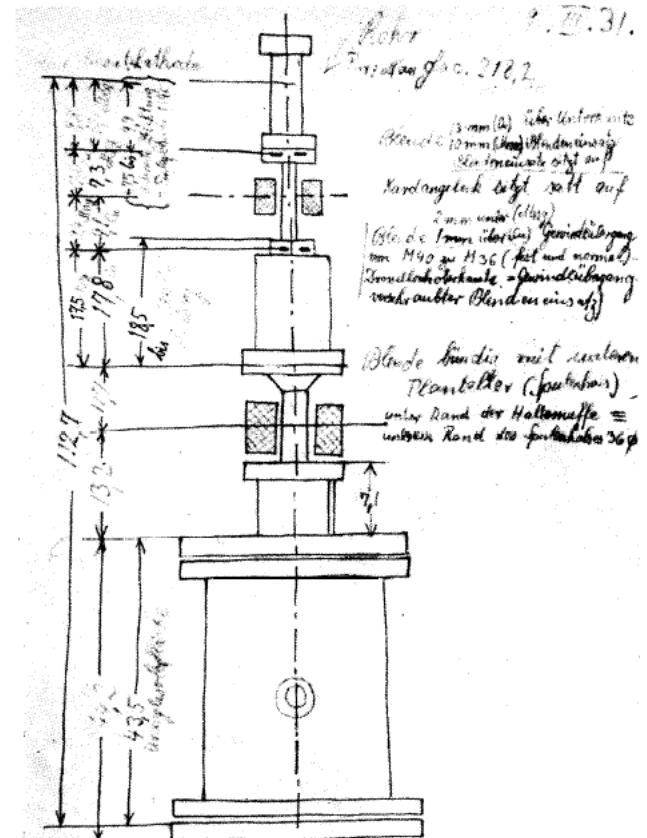


Fig. 2: Sketch by the author (9 March 1931) of the cathode ray tube for testing one-stage and two-stage electron-optical imaging by means of two magnetic electron lenses (electron microscope) [8].

In situ Ion Irradiation TEM Facility

Collaborators: D.L. Buller & J.A. Scott

200 keV JEOL-2100

Stages:

- Heating
- LN cooling
- 3 × Mechanical loading
- Gas/liquid flow
- 2 × Tomography

10 kV
Colutron

6 MV
Tandem

Bending
magnet to
mix beams

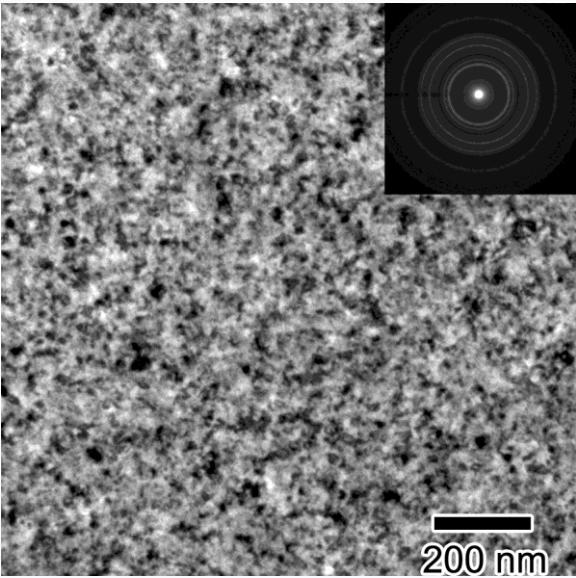
Pre-TEM *ex situ* irradiation
chamber



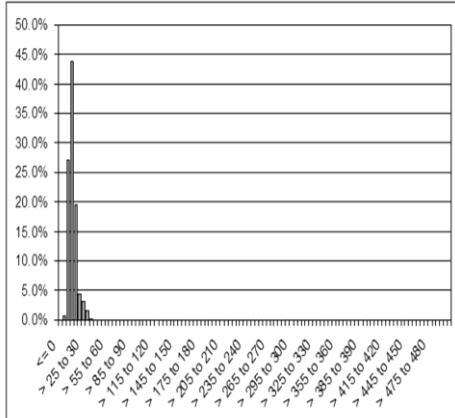
Thermal Stability of Nanocrystalline Materials

Collaboration with: J.A. Knapp, D.M. Follstaedt, and I.M. Robertson

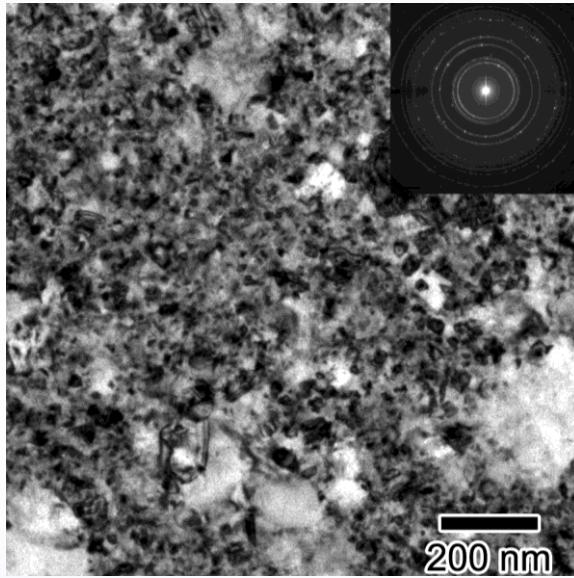
Nanograined



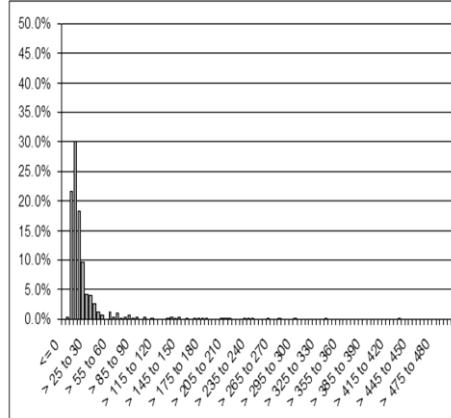
80 nm-thick As-deposited on Si



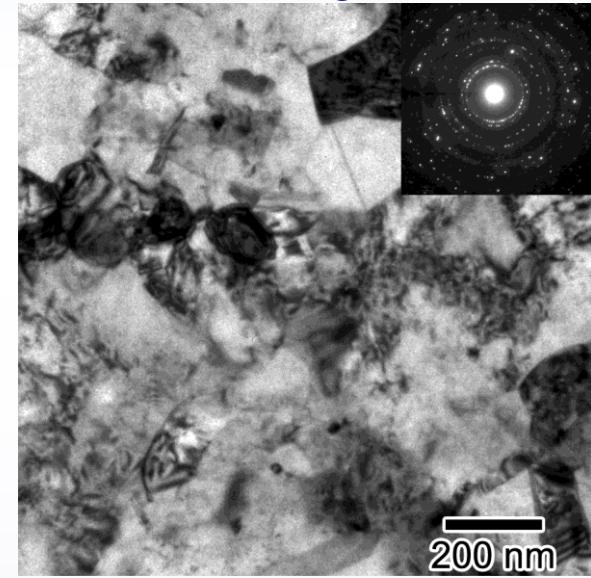
Bimodal



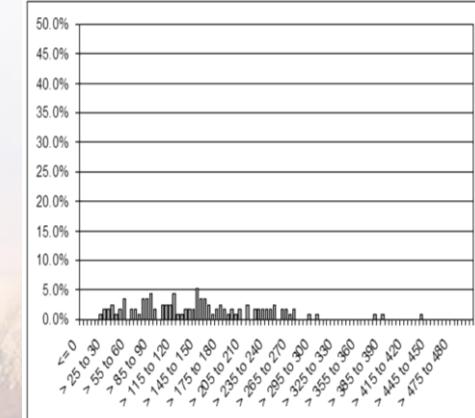
1 hr at 275 ° C



Ultra-fine grained



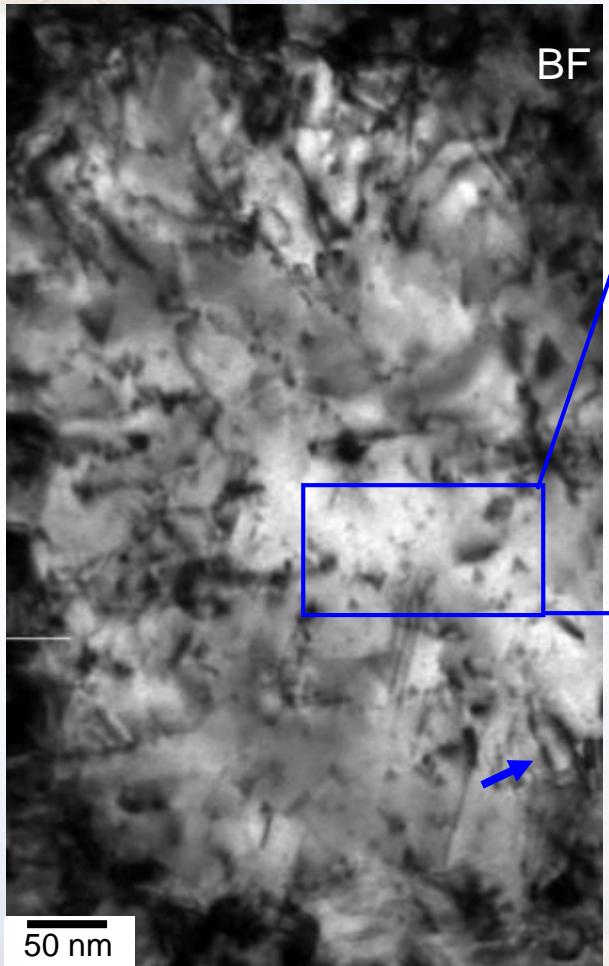
1 hr at 375 ° C



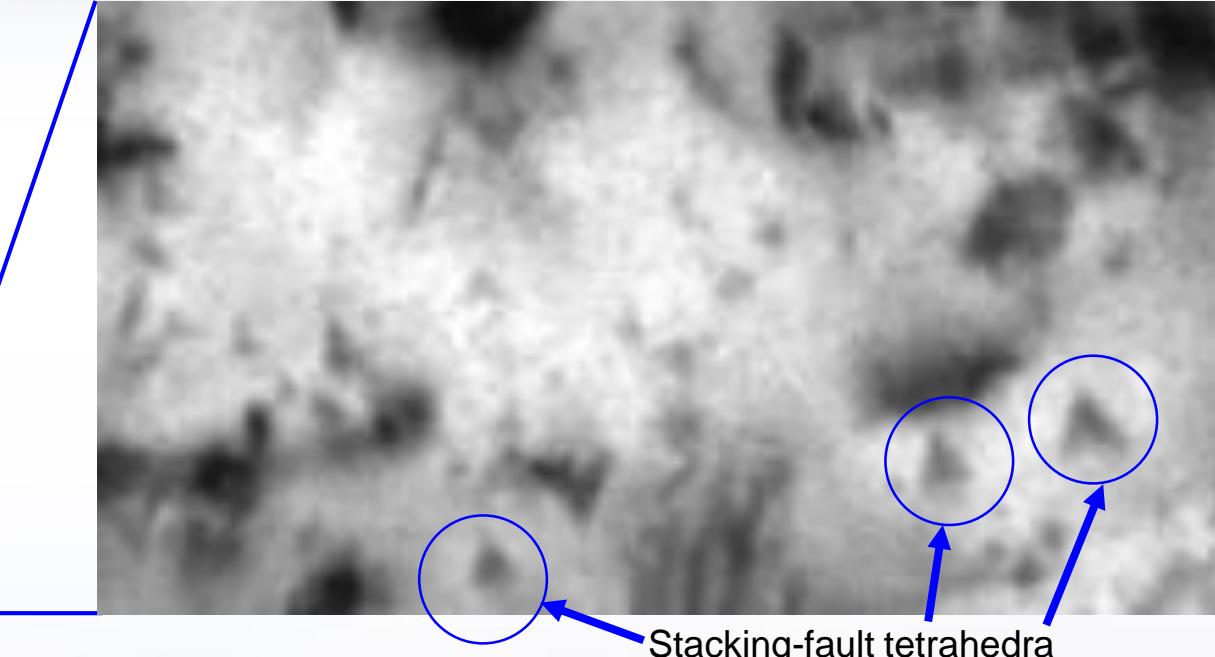
A few select grains grow at the expense of the remaining matrix

A Variety of Unexpected Defect Structures in Ni

Collaboration with: J.A. Knapp, D.M. Follstaedt, and I.M. Robertson

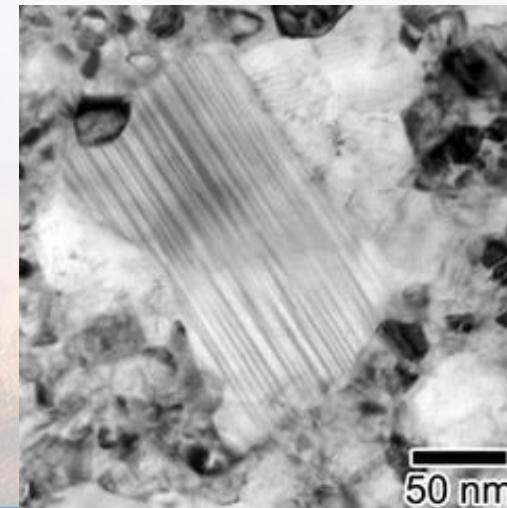


230 nm thick PLD Ni Annealed at 225 °C for 14 hrs.



Multitude of defects in annealed PLD Ni

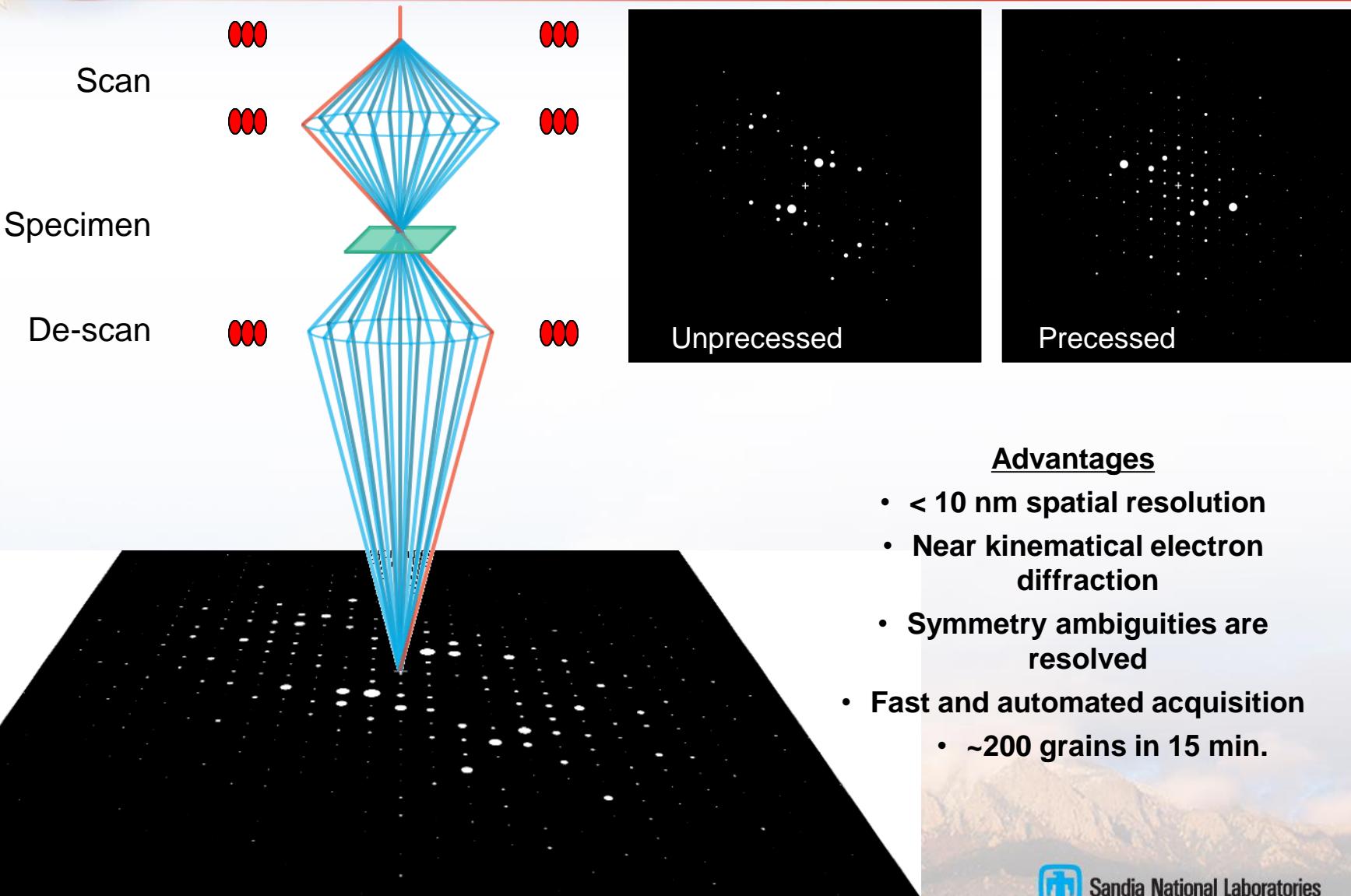
- SFT at temperature
- Stable microstructure for over 15 months
- SFT not due to irradiation, quenching, high strain rate
- SFT are theorized to be formed by rapid grain growth through the high free-volume at the initial grain boundaries



50 nm

Precession Electron Diffraction (PED) Microscopy

Collaborators: K.J. Ganesh, S. Rajasekharan, & P.J. Ferreira



Advantages

- < 10 nm spatial resolution
- Near kinematical electron diffraction
- Symmetry ambiguities are resolved
- Fast and automated acquisition
 - ~200 grains in 15 min.



In Situ Irradiation of Nanocrystalline Au

- Au foil during bombardment with 10 MeV Si³⁺
- ~22 s of 4000s total experiment time

Locations of single ion strikes and resulting microstructural change captured.

In situ ion irradiation
TEM: 10 MeV Si into
nanocrystalline Au.

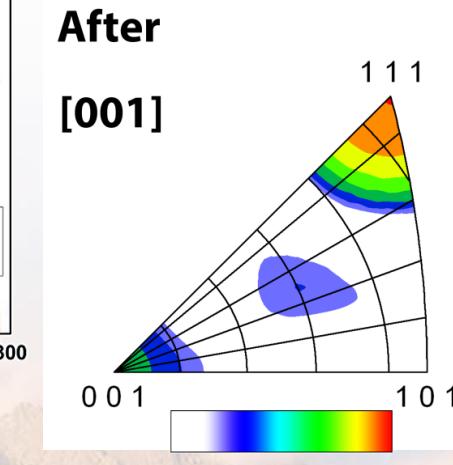
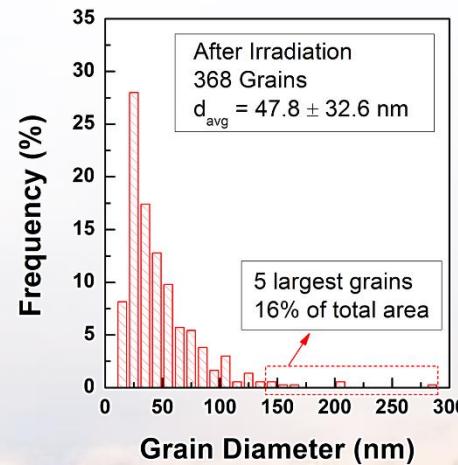
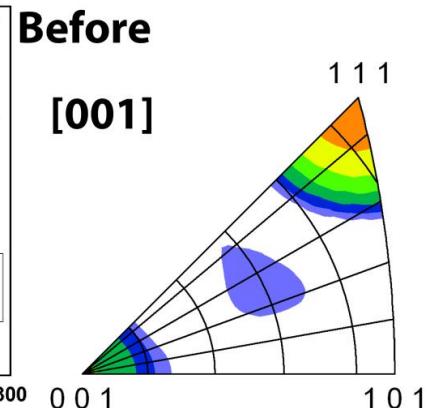
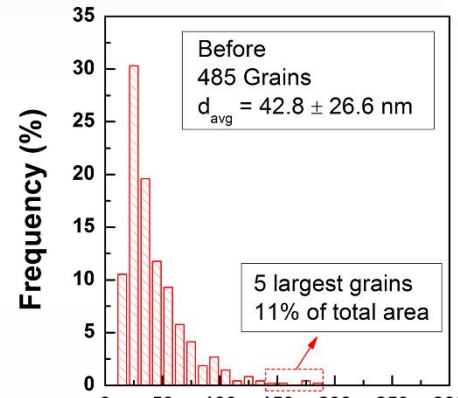
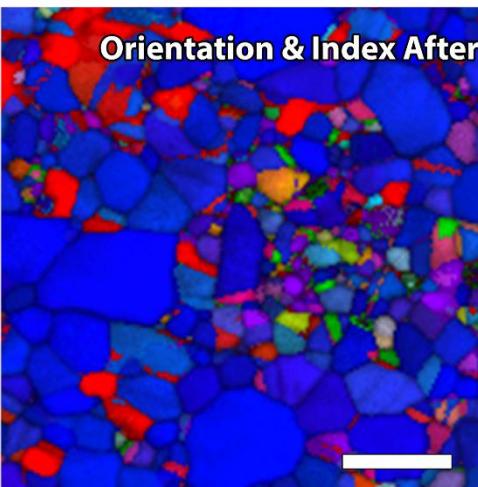
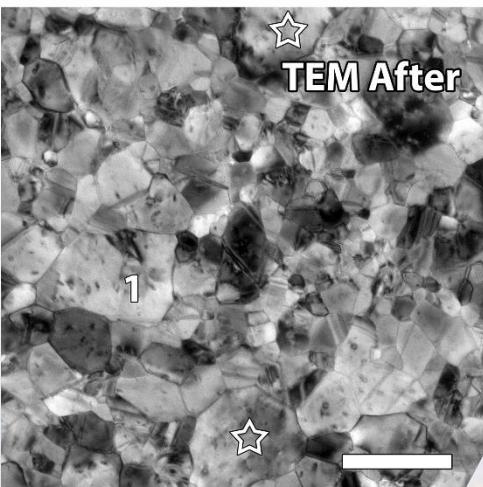
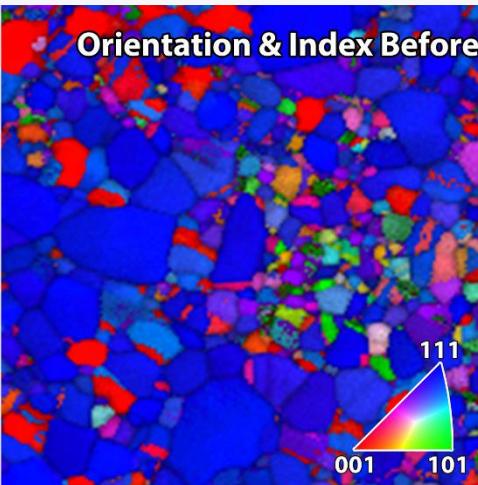
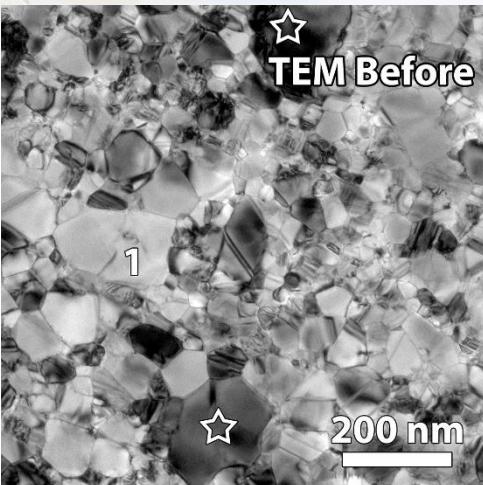
Playback at 2 × real time.

2 × real time



Quantifying Stability of Nanocrystalline Au during 10 MeV Si Ion Irradiation

Collaborators: D.C. Bufford, F. Abdeljawad, & S.M. Foiles



Increasing Intensity



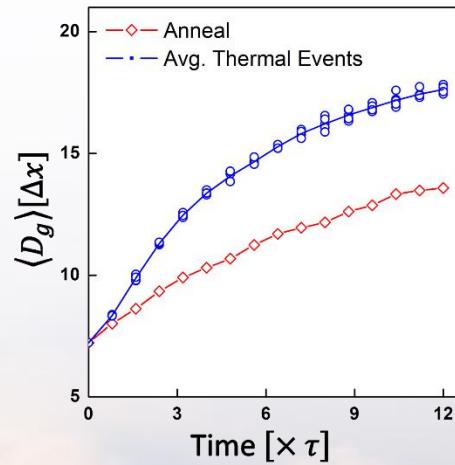
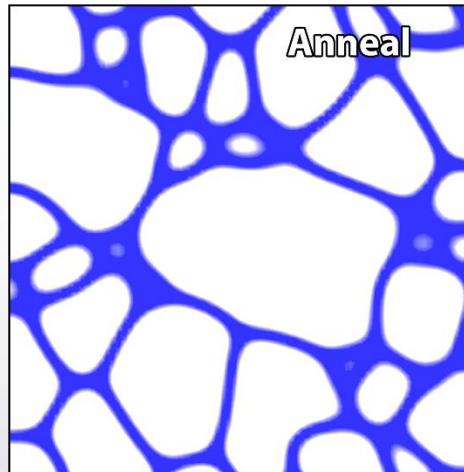
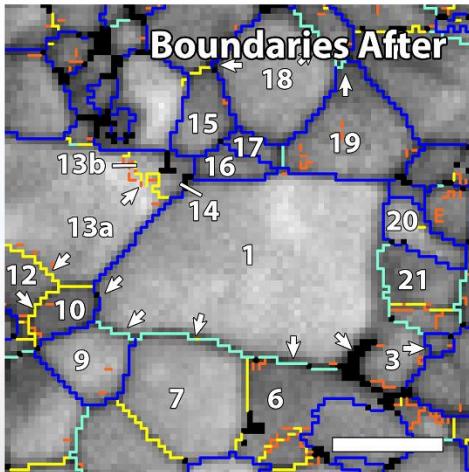
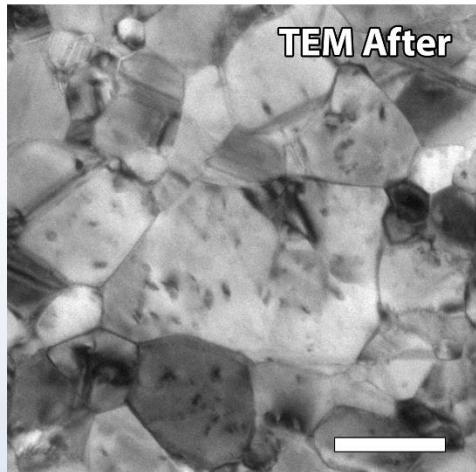
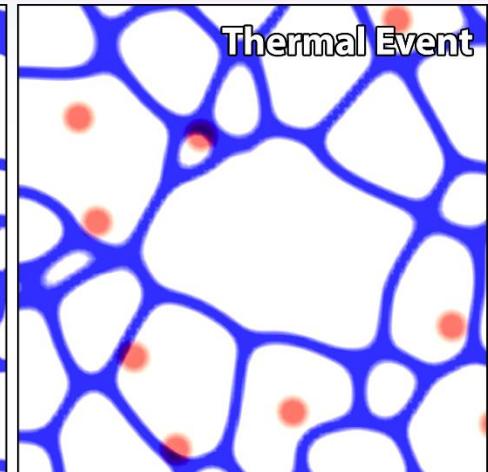
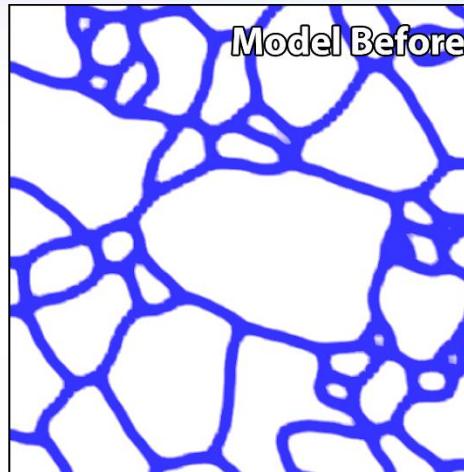
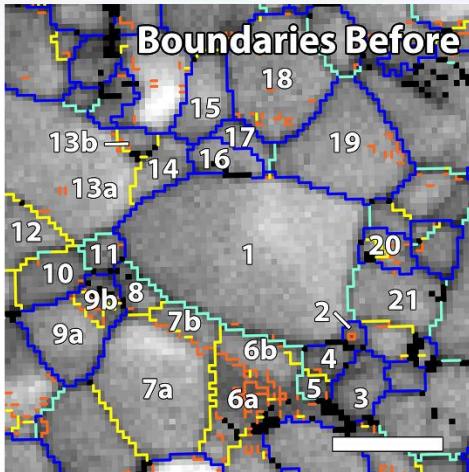
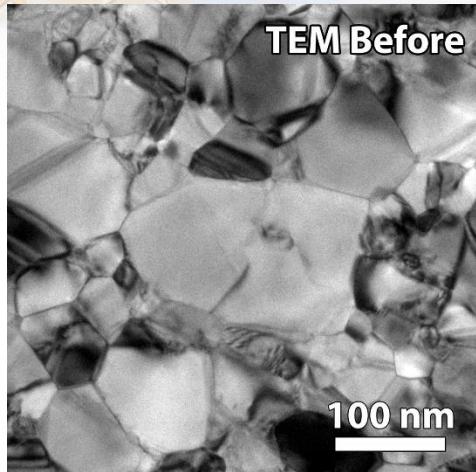
Any texture or grain boundary evolution can be directly observed and quantified



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Direct Comparison to Mesoscale Modeling

Collaborators: D.C. Bufford, F. Abdeljawad, & S.M. Foiles



3° $\leq \varphi < 15^\circ$
15° $\leq \varphi < 30^\circ$
30° $\leq \varphi$

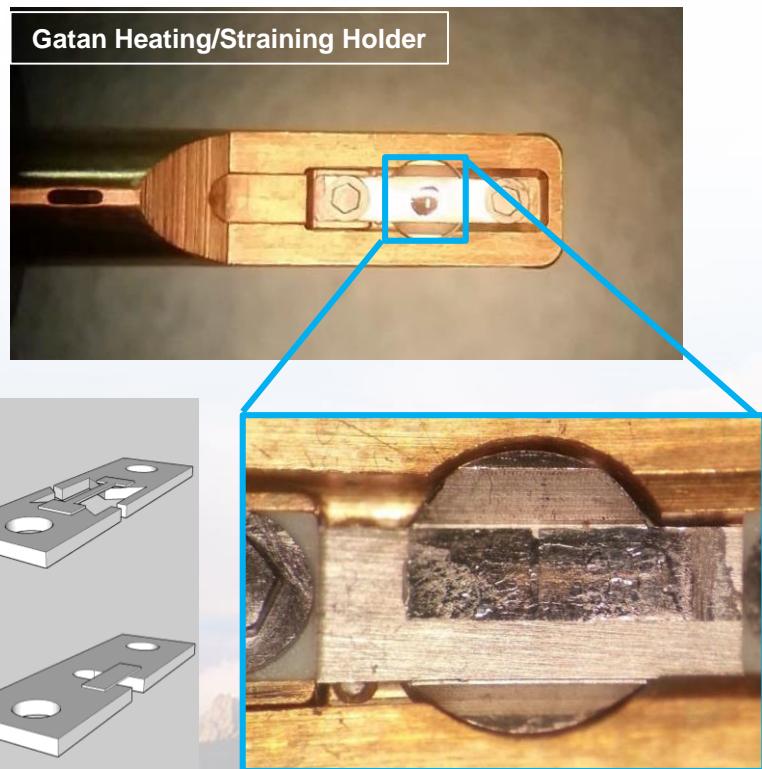
Because of the matching length scale, the initial microstructure can serve as direct input to either MD or mesoscale models & subsequent structural evolution can be directly compared.



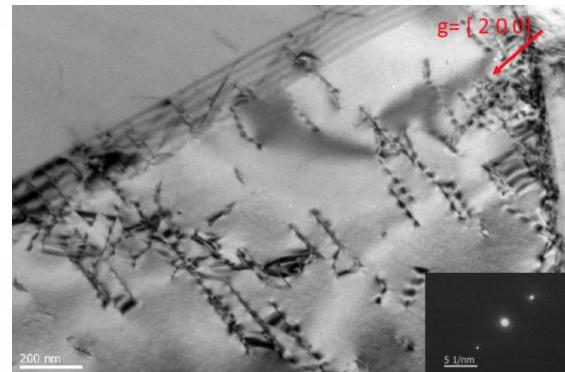
In situ Qualitative Mechanical Testing

In Situ Straining TEM Holder

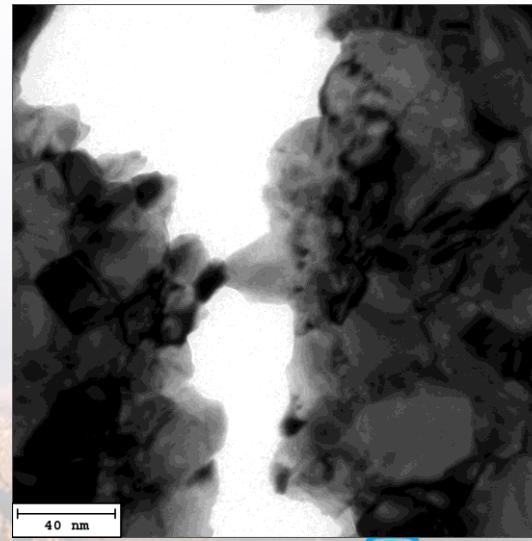
- Minimal control over displacement
- No out-of-box force information
- Ability to observe dislocation-GB interactions



Observe dislocation and pile-up interactions as function of GB character



Observe deformation mechanisms in nanocrystalline metals during tensile straining



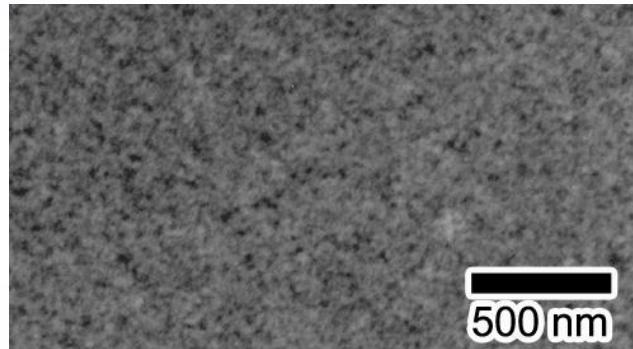
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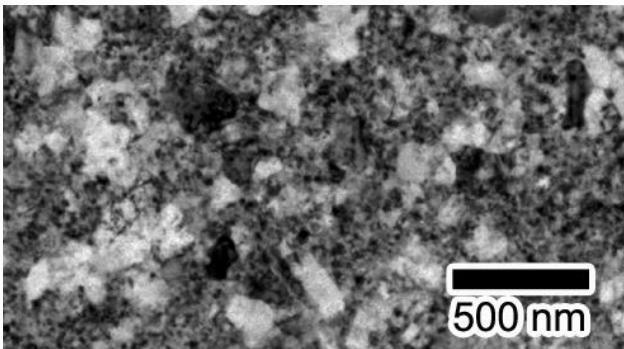
Failure Analysis of Strained PLD Ni

Collaboration with: J.A. Knapp, D.M. Follstaedt, and I.M. Robertson

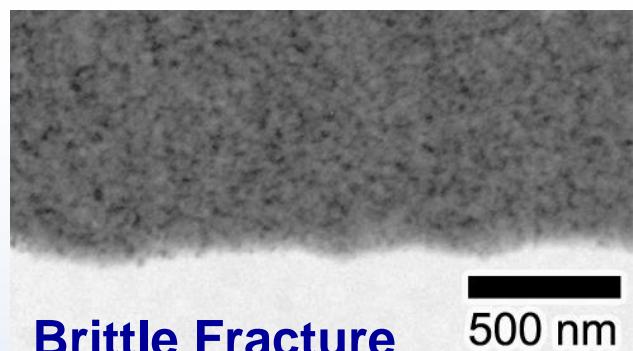
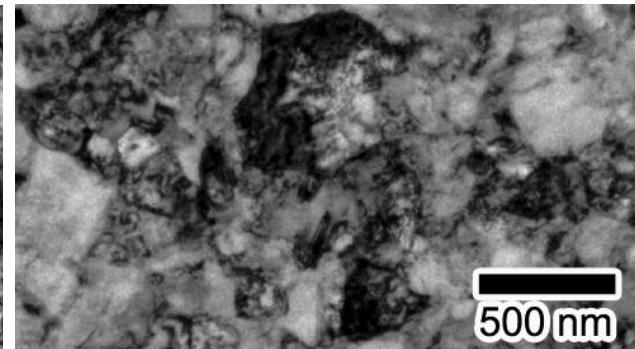
Nanograined



Bimodal

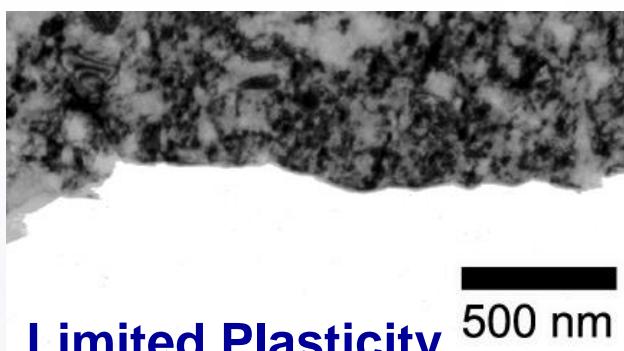


Ultra-fine grained



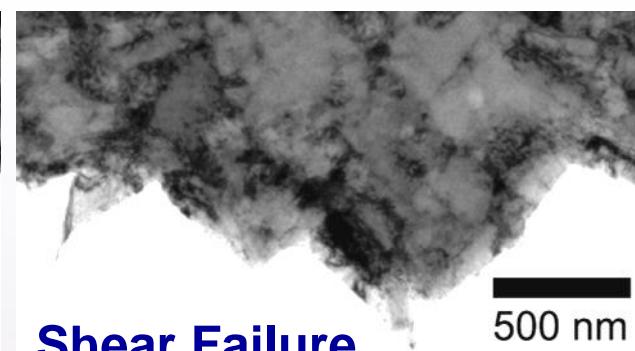
Brittle Fracture

500 nm



Limited Plasticity

500 nm



Shear Failure

500 nm

- No observation of global plasticity

- Dislocation pile-up
- Local shear

- Shear teeth
- Dislocation structure

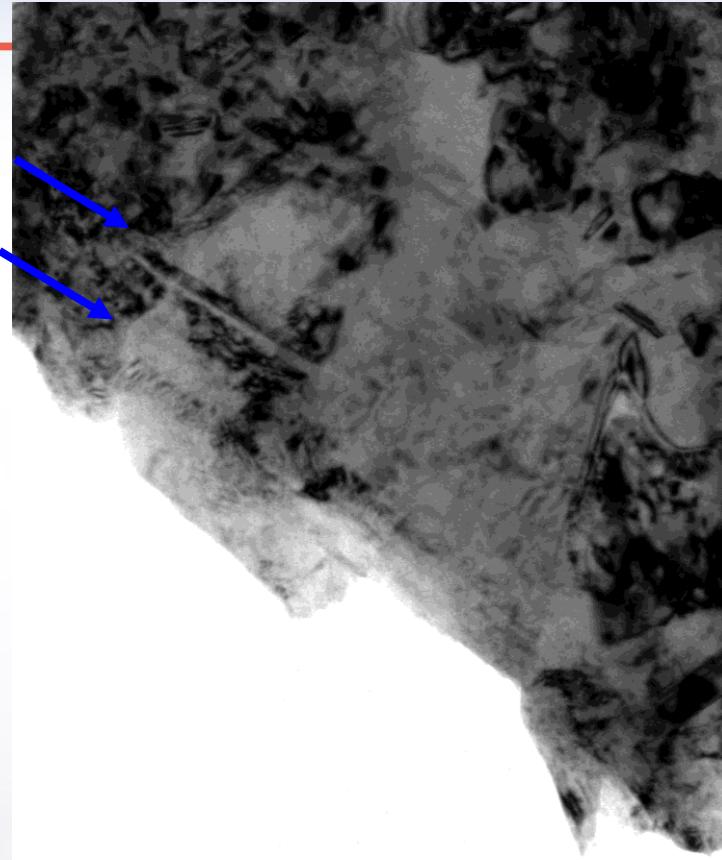
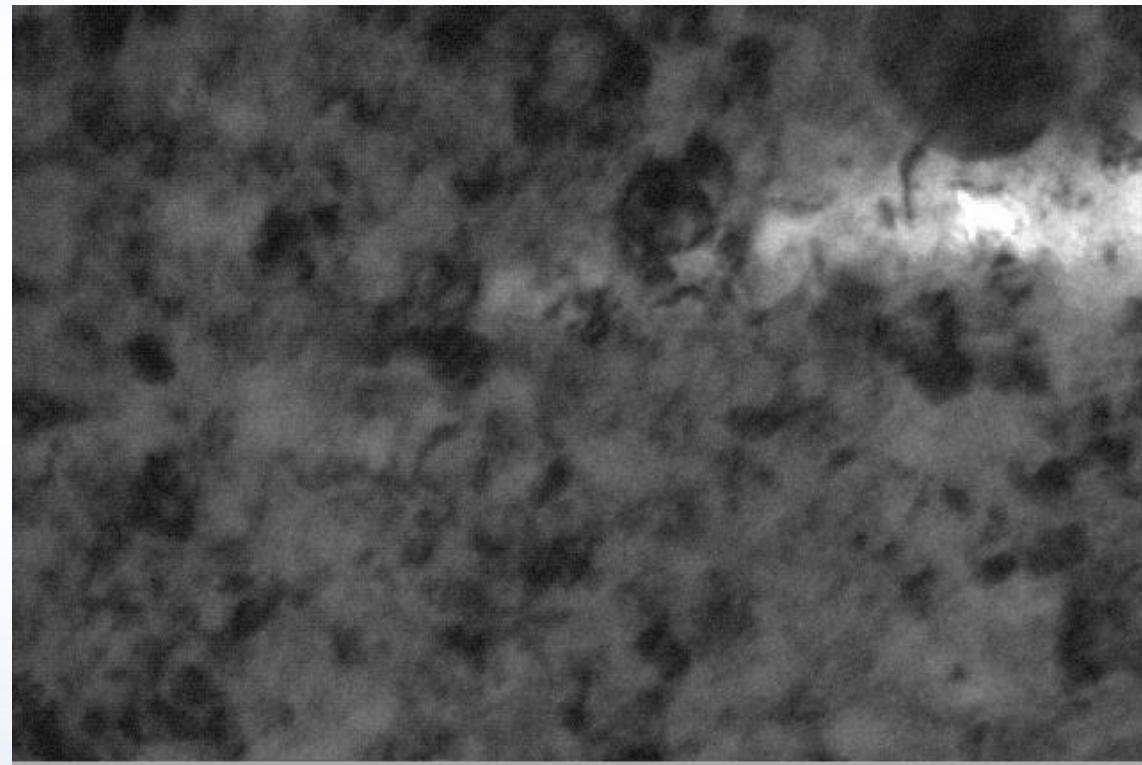
Fracture surfaces provide insight to deformation processes



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Deformation and Failure in Bimodal Ni

80 nm-thick PLD Ni annealed at 275 °C



Throughout the film

- Elastic strain
- Limited dislocation slip

In the plastic zone

- Extensive dislocations slip
- Twinning

At Crack Tip

- Necking
- Grain agglomeration

We have some insight into the unique thermal and mechanical mechanisms and properties. What is the initial nanostructure that causes this?



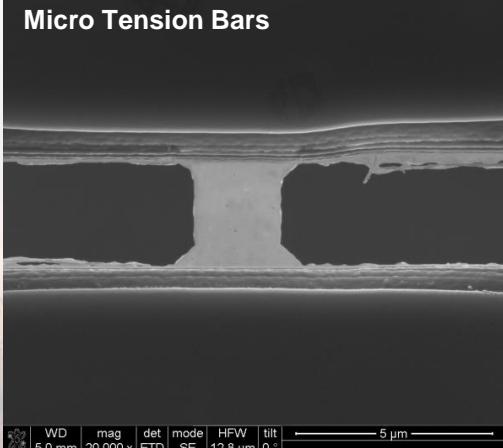
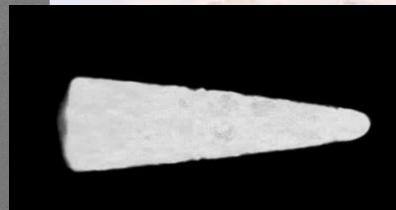
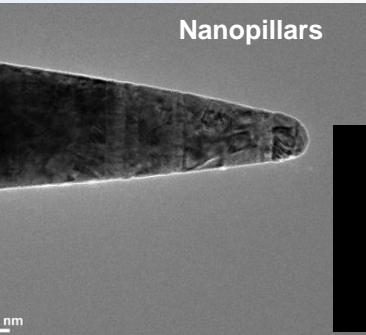
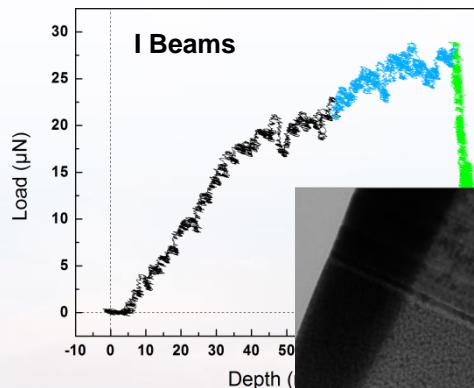
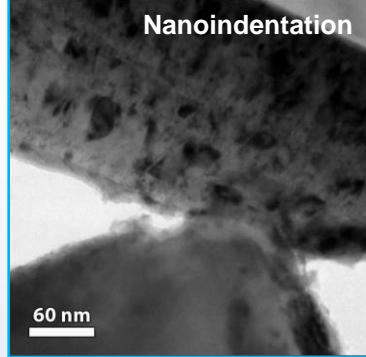
In situ Quantitative Mechanical Testing

Contributors: J. Sharon, B. L. Boyce, C. Chisholm, H. Bei, E.P. George, P. Hosemann, A.M. Minor, & Hysitron Inc.

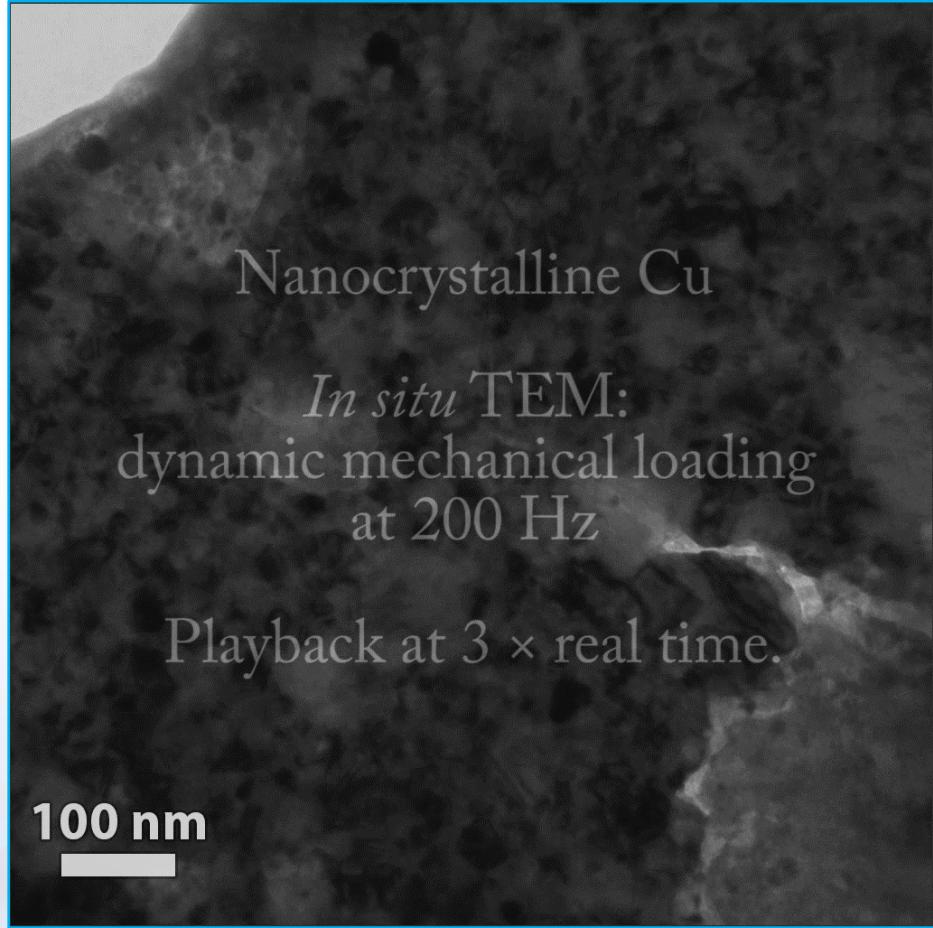


Hysitron PI95 *In Situ* Nanoindentation TEM Holder

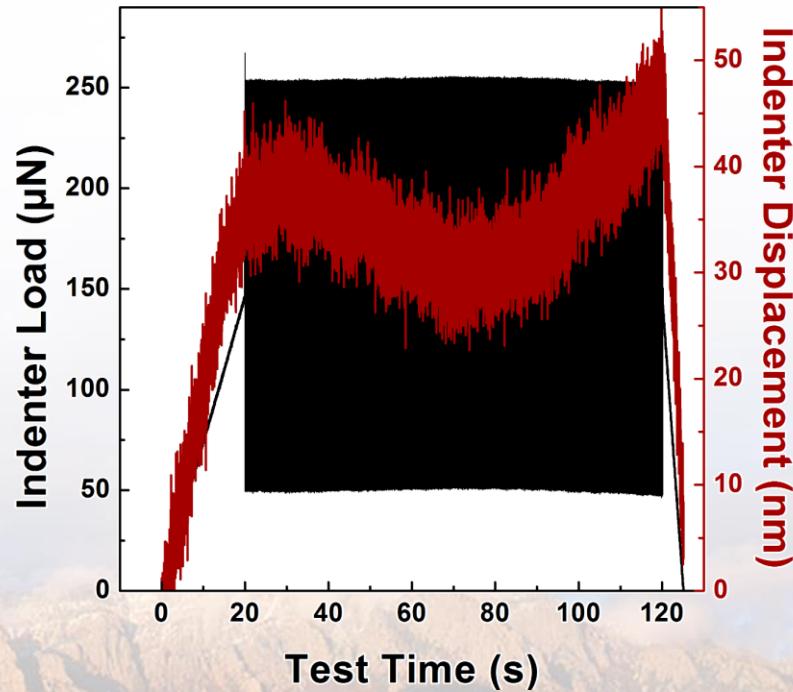
- Sub nanometer displacement resolution
- Quantitative force information with μN resolution
- **Concurrent real-time imaging by TEM**



High Cycle Fatigue: Experimental Observations



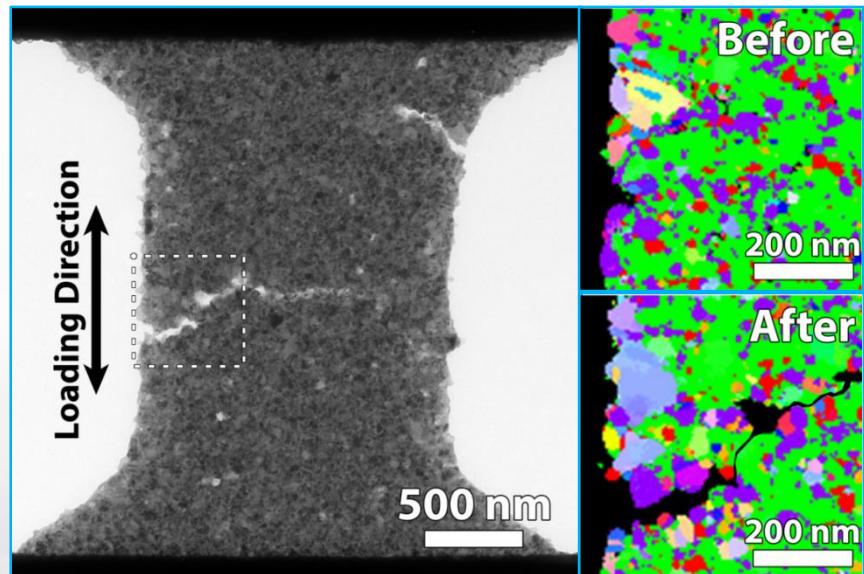
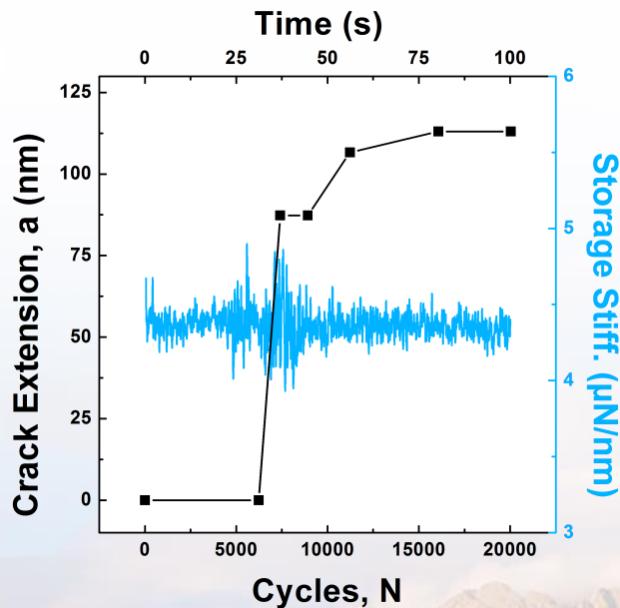
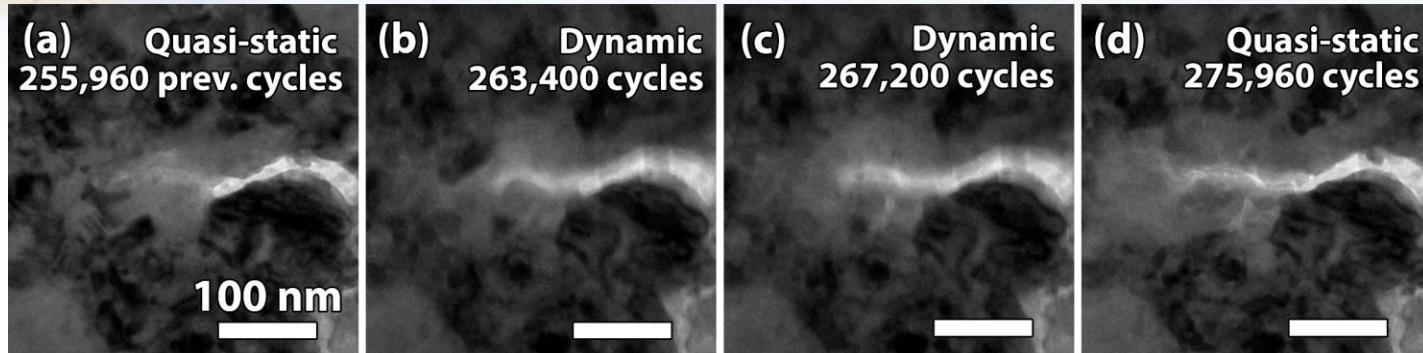
- Cyclic loading at 200 Hz
- Structural change at crack tip captured



First high cycle fatigue experiments with continuous observation by TEM!



High Cycle Fatigue: Crack Growth Quantified

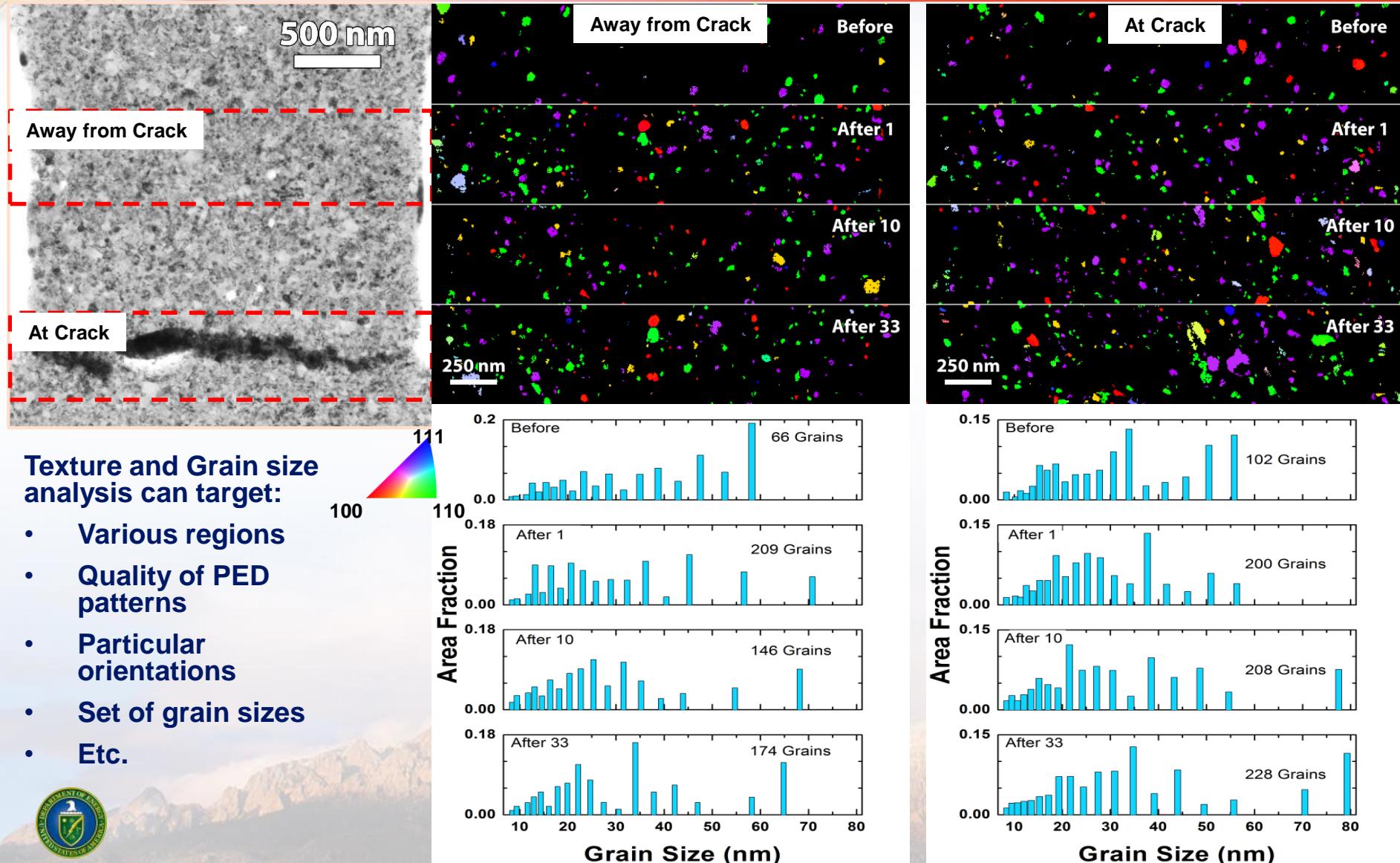


- Picometer-per-cycle measured crack growth rate
- Evidence of fatigue-induced grain growth.



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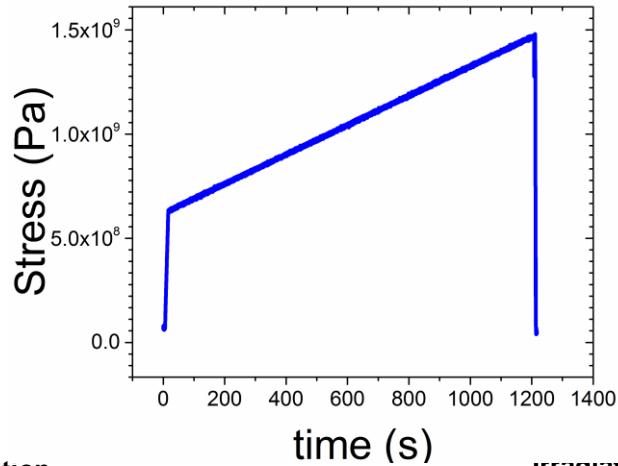
Comparison of Grain Size Near and Far from the Fracture Surface



Irradiation Creep (4 MeV Cu³⁺ 10⁻² DPA/s)

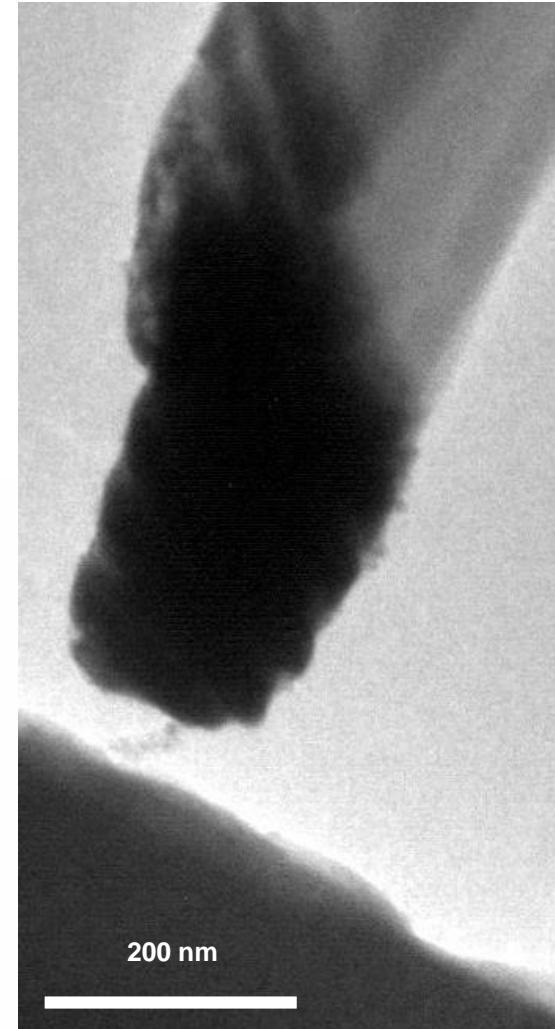
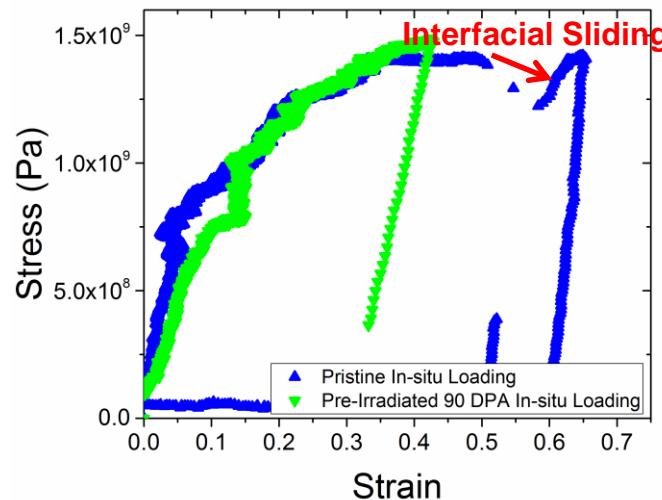
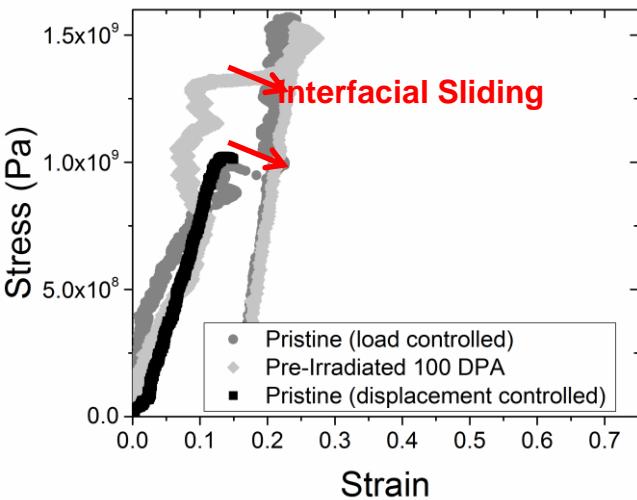
Controlled Loading Rate Experiments

50 nm Cu-W multilayer
20 Min



No Irradiation
(Loading rate 0.6 Mpa s⁻¹)

irradiation Creep
(Loading rate 0.6 Mpa s⁻¹)



In situ TEM Tensile Testing of Ion Irradiated Mo-alloy Wires

Contributors: C. Chisholm, H. Bei, E.P. George, P. Hosemann, & A.M. Minor

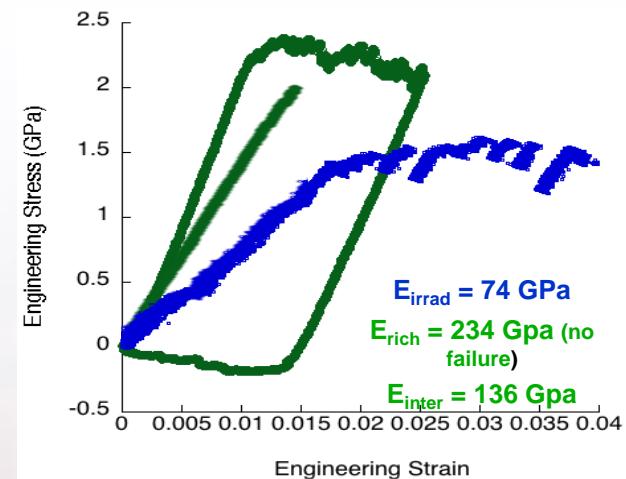
$1.92 \times 10^{18} \text{ He}^{1+}/\text{cm}^2$

$1.04 \times 10^{15} \text{ Ni}^{2+}/\text{cm}^2$

0.1 μm

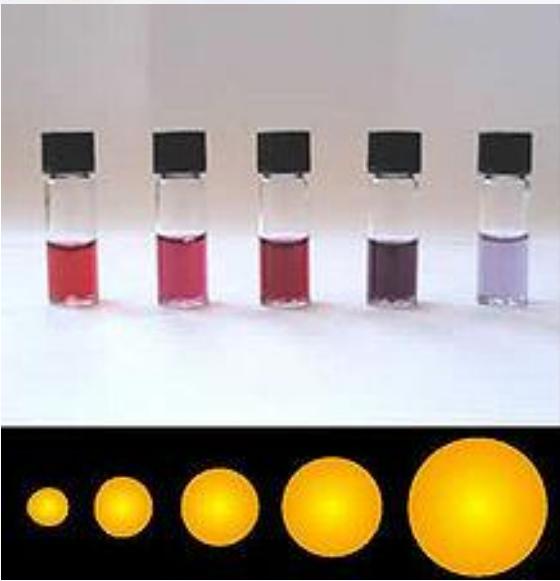
Intermediate
dislocation
density

0.5 μm



Dislocation structure and
subsequent deformation and failure
mechanisms through controlled ion
beam modification

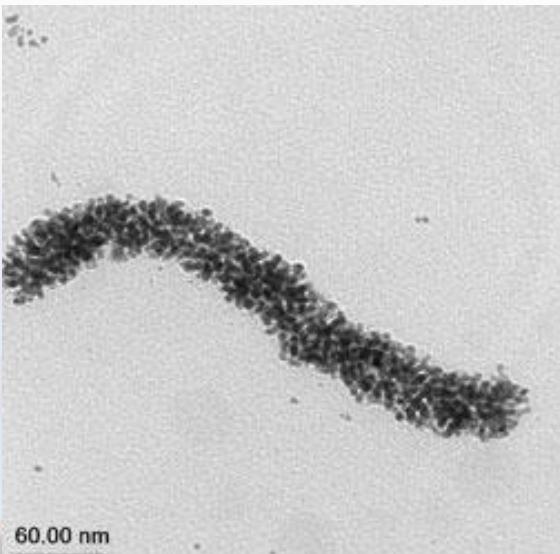
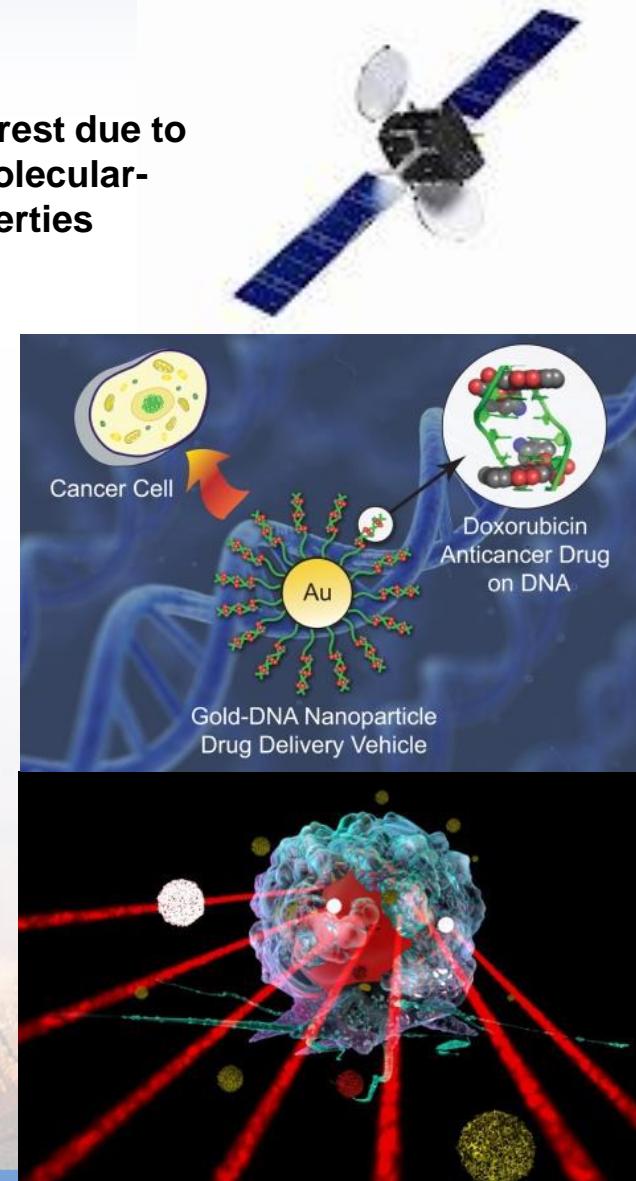
Nanoparticles in Extreme Environments



Au nanoparticles (Au NP) are of interest due to their unique optical, electronic, molecular-recognition, and catalytic properties

- Photothermal and laser ablation cancer therapies – concentrate IR into heat
- Tumor targeting and drug carriers – enhance dose delivery
- Catalysts for air-pollution control

Do the unique properties of the NP withstand extreme radiation?



Cumulative Effects of Ion Irradiation as a Function of Ion Energy and Au Particle Size

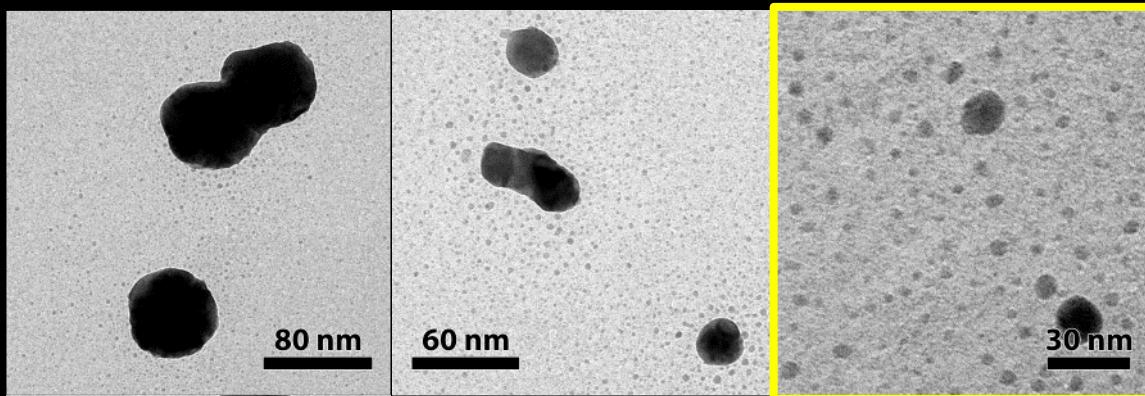
60 nm

20 nm

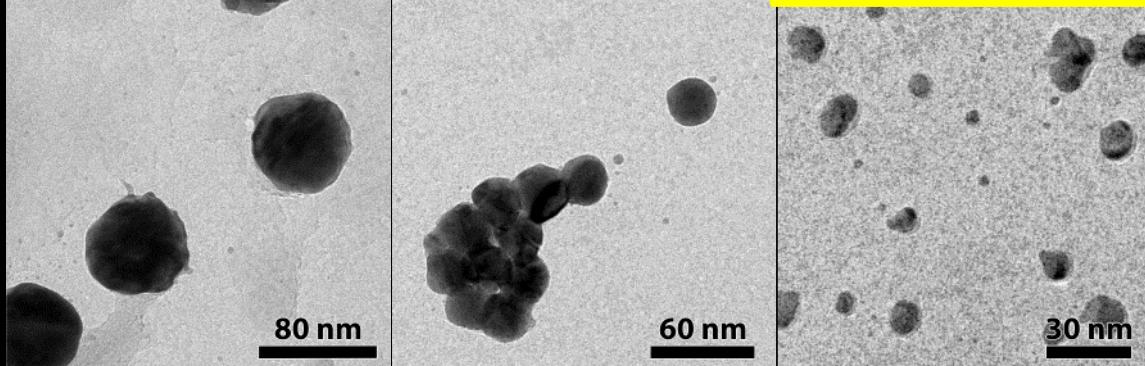
5 nm

Collaborator: D.C. Bufford

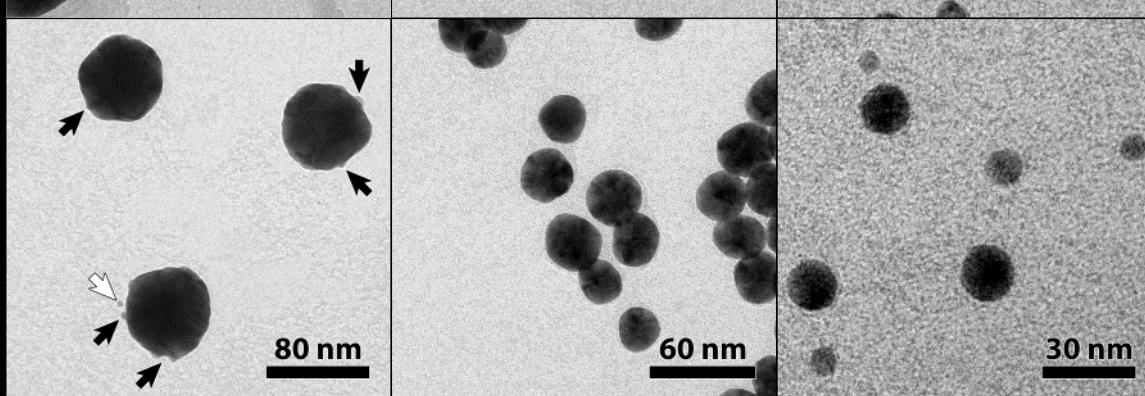
46 keV Au¹⁻
 $3.4 \times 10^{14} / \text{cm}^2$



2.8 MeV Au⁴⁺
 $4 \times 10^{13} / \text{cm}^2$



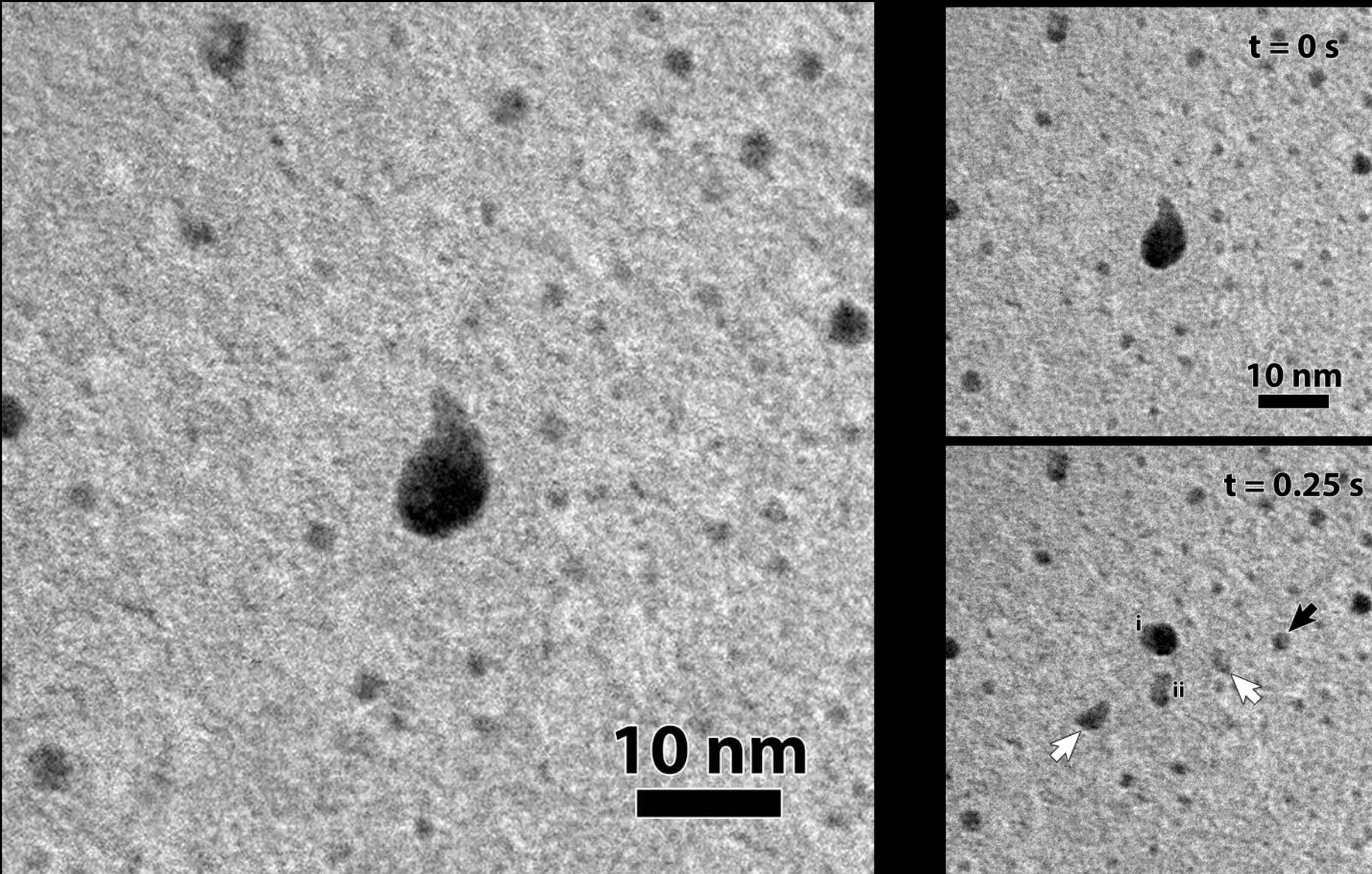
10 MeV Au⁸⁺
 $1.3 \times 10^{12} / \text{cm}^2$



Particle and ion energy dictate the ratio of sputtering, particle motion, particle agglomeration, and other active mechanisms

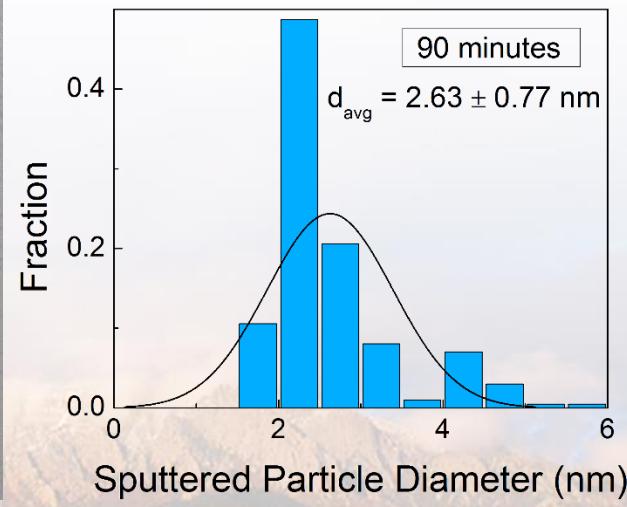
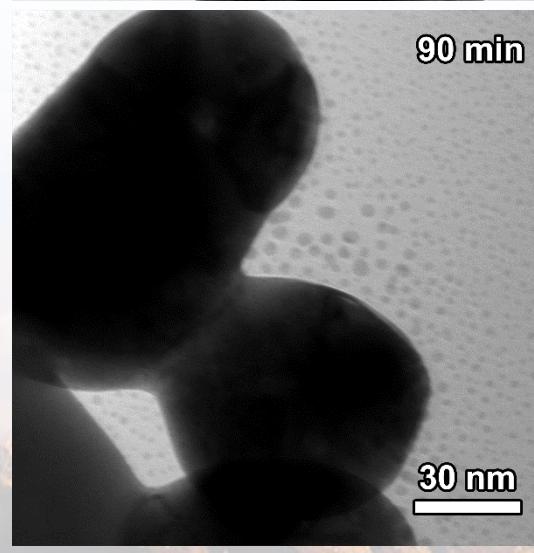
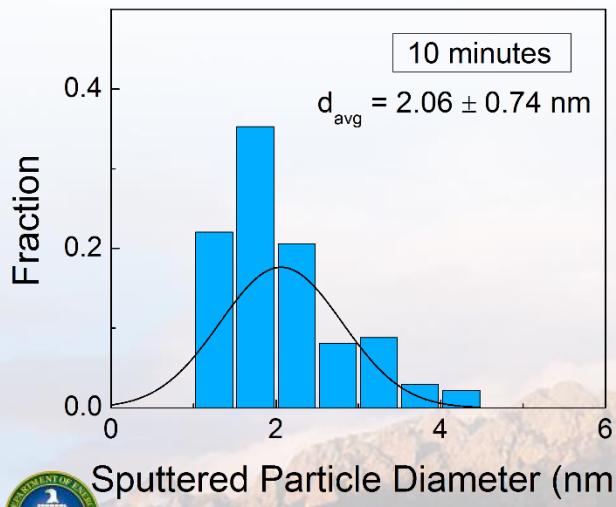
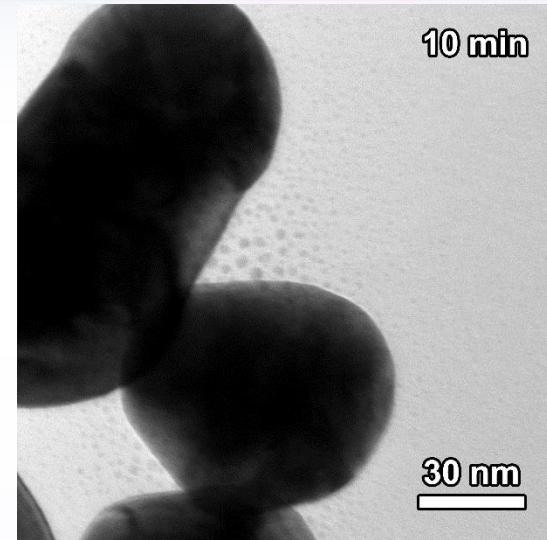
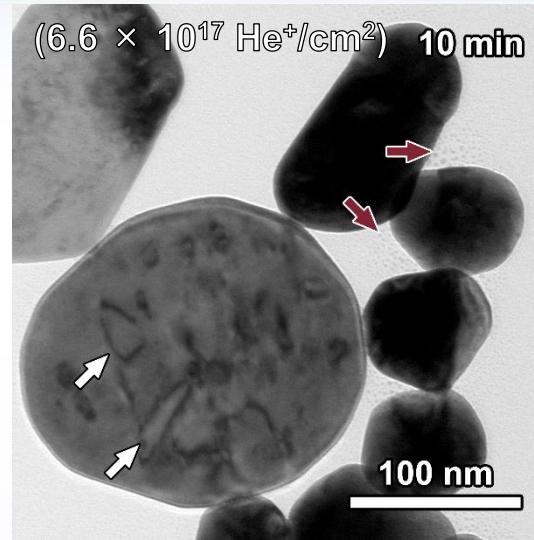
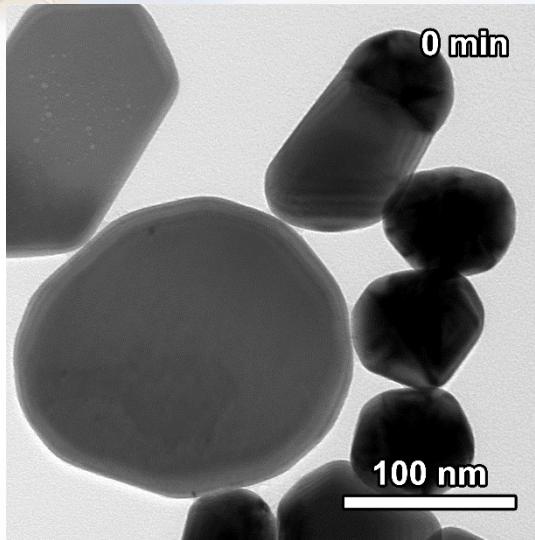
Single Ion Effects with 46 keV Au¹⁻ ions: 5 nm

Collaborator: D.C. Bufford



Formation of Dislocation Loops & Sputtered Particles due to He implantation

Collaborators: D.C. Bufford, S.H. Pratt & T.J. Boyle



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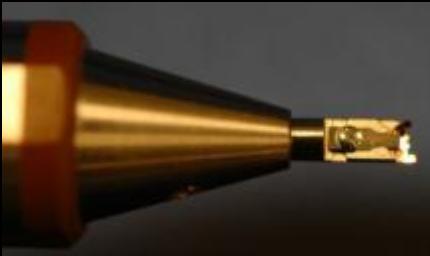
Advanced Microscopy Techniques Applied to Nanoparticles in Radiation Environments

Collaborators: S.M. Hoppe & T.J. Boyle

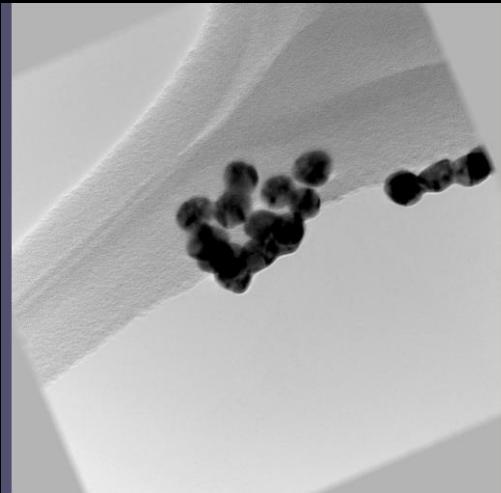
In situ Ion Irradiation TEM (I³TEM)



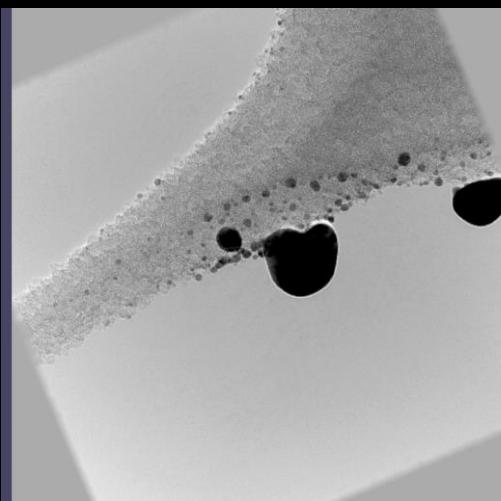
Hummingbird
tomography stage



Aligned Au NP tilt series -
unirradiated



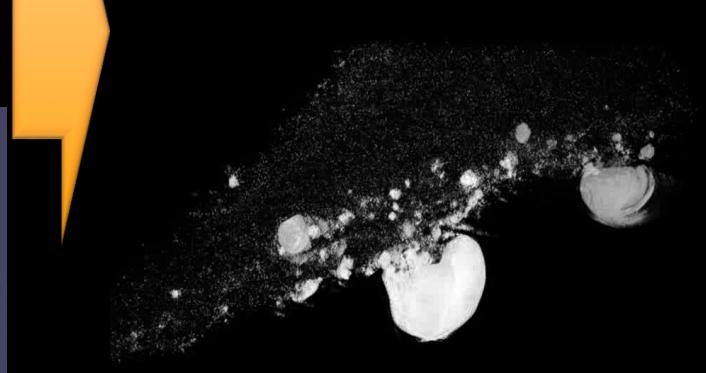
Aligned Au NP tilt series -
irradiated



Unirradiated Au NP model



Irradiated Au NP model



The application of advanced
microscopy techniques to
extreme environments provides
exciting new research directions

Radiation Tolerance is Needed in Advanced Scintillators for Non-proliferation Applications

Contributors: S.M. Hoppe, B.A. Hernandez-Sanchez, T. Boyle



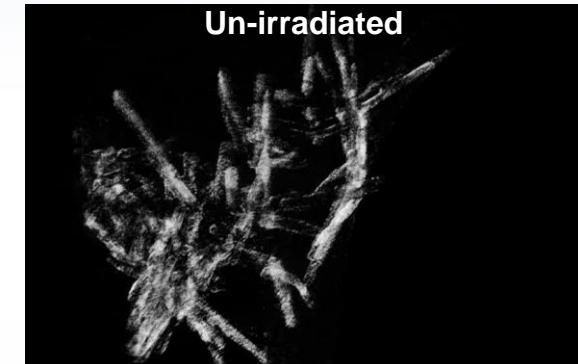
In situ Ion Irradiation TEM (I³TEM)



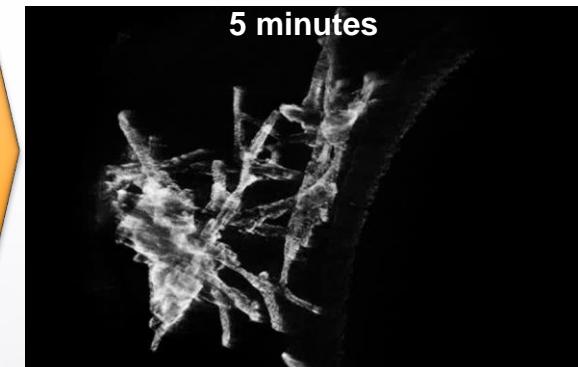
Hummingbird tomography stage



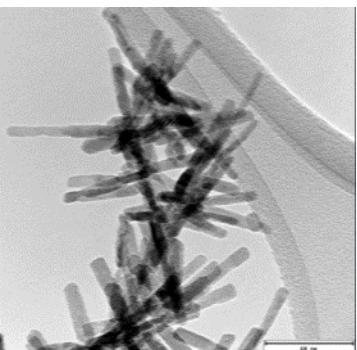
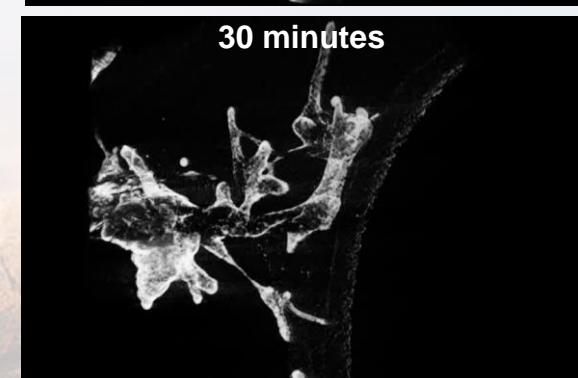
Tomography of Irradiated CdWO₄:
3 MeV Cu³⁺ at ~30 nA



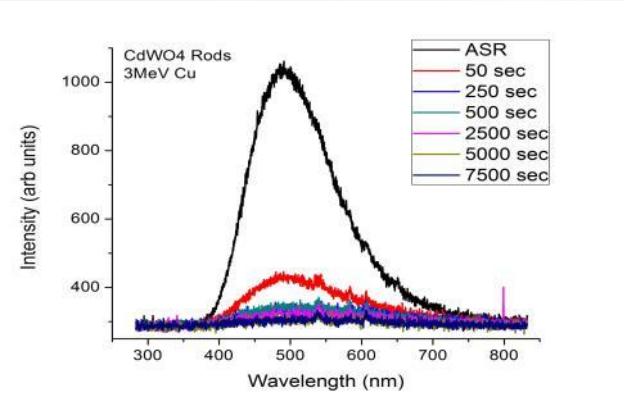
5 minutes



30 minutes



High-Z nanoparticles (CdWO₄) are promising, but are radiation sensitive



Can We Gain Insight into the Corrosion and Biofouling Process through *In situ* TEM?

Contributors: D. Gross, J. Kacher, I.M. Robertson & Protochips, Inc.

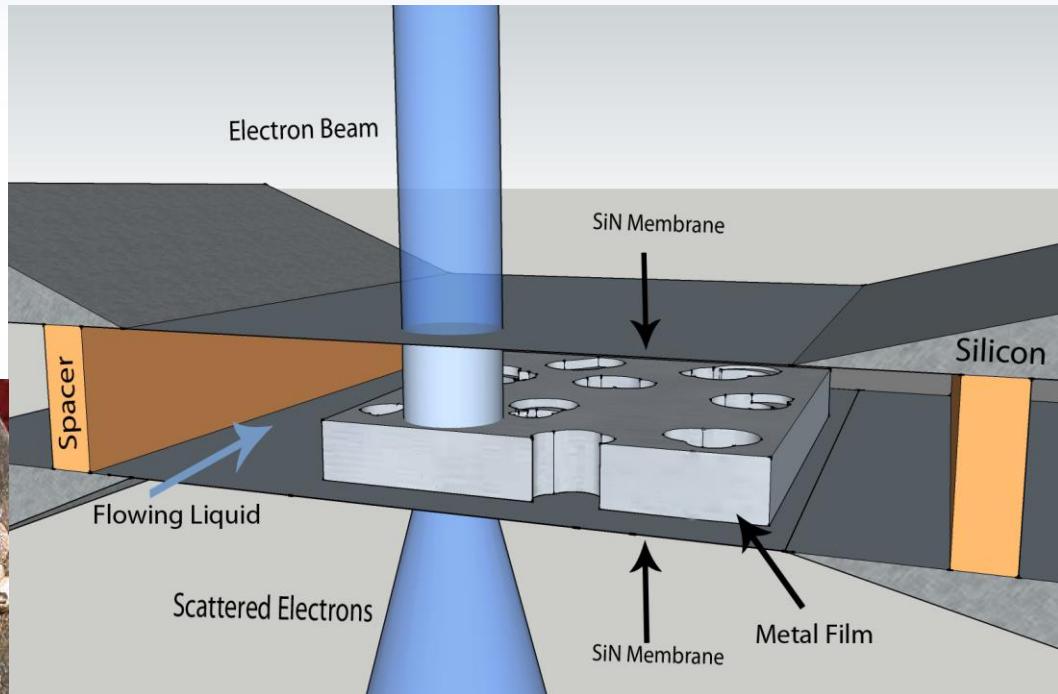


Biocorrosion

Drag

Introduces corrosive species

Decreased performance



Microfluidic Stage

- Mixing of two or more channels
- Continuous observation of the reaction channel
- Chamber dimensions are controllable
- Films can be directly deposited on the SiN membrane

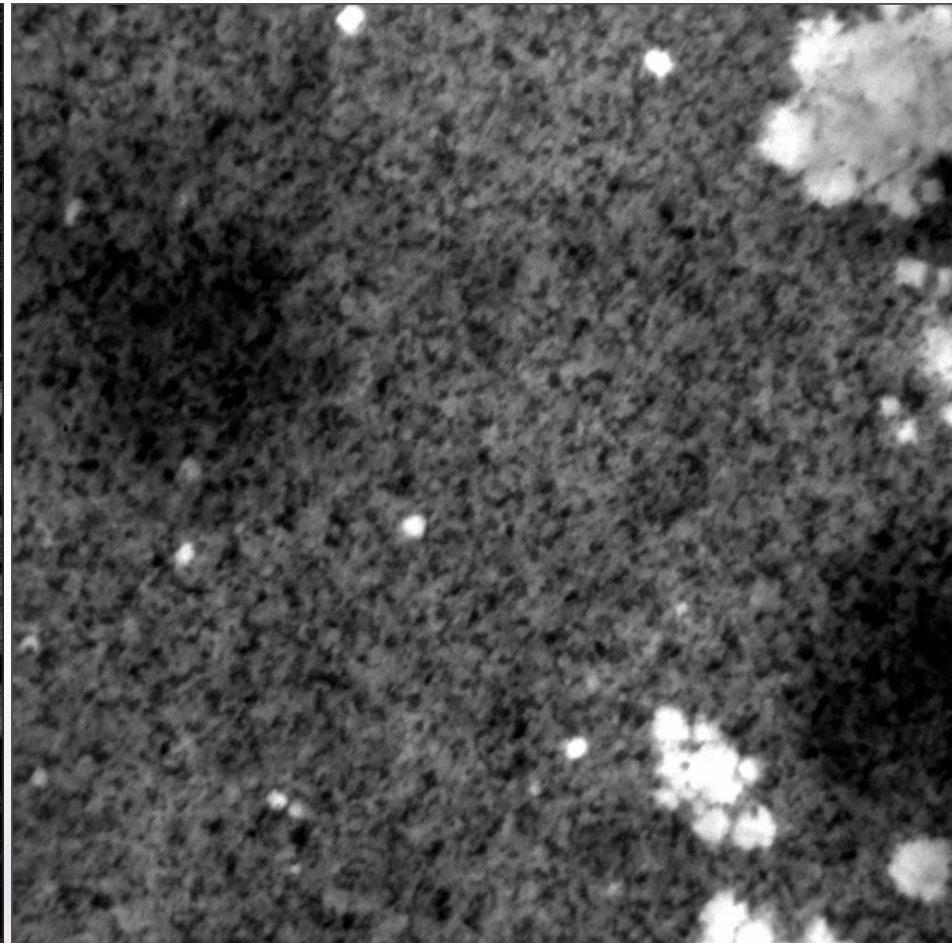
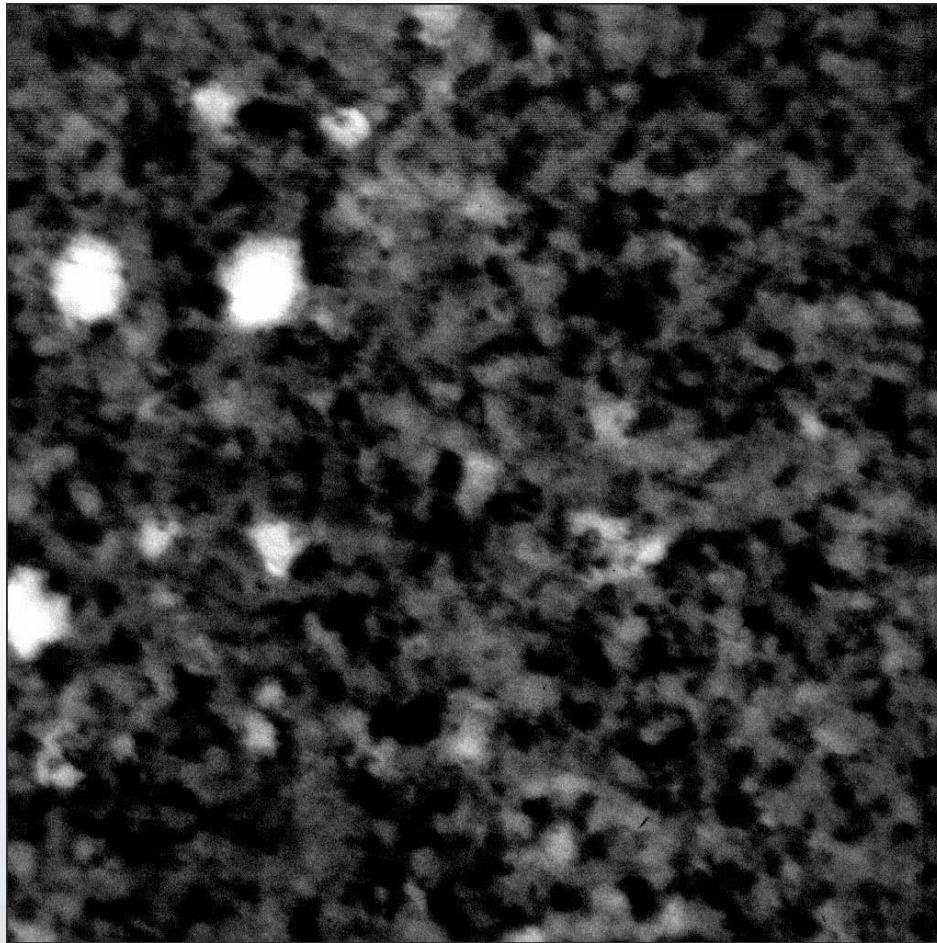


More than \$5.7 billion is spent annually on control and prevention



Acetic Acid Corroding Nanocrystalline Iron

Collaborators: D. Gross, J. Kacher, & I.M. Robertson



Pitting mechanisms during dilute flow of acetic acid over 99.95% nc-PLD Fe involves many grains.

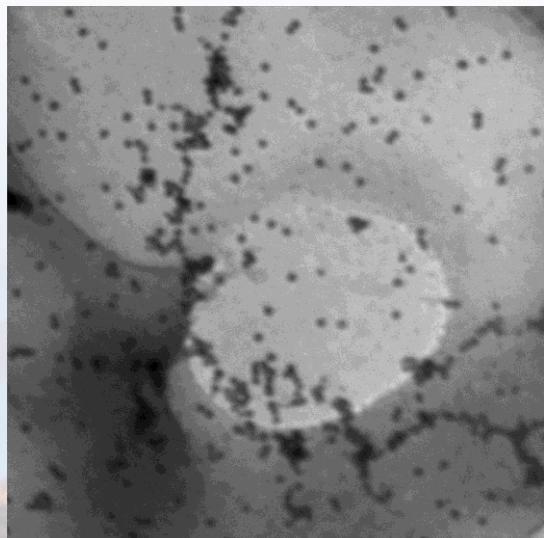
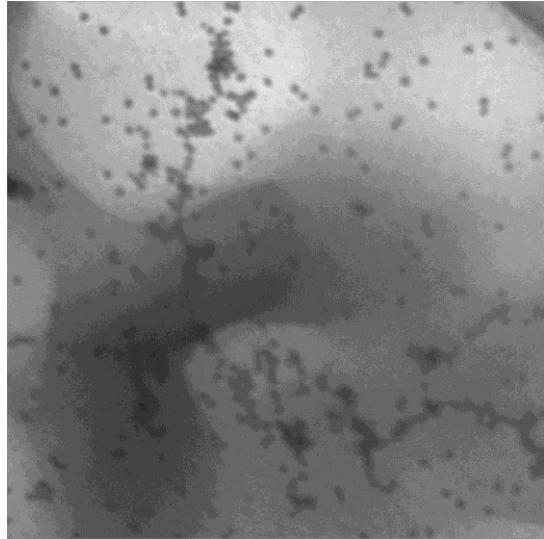


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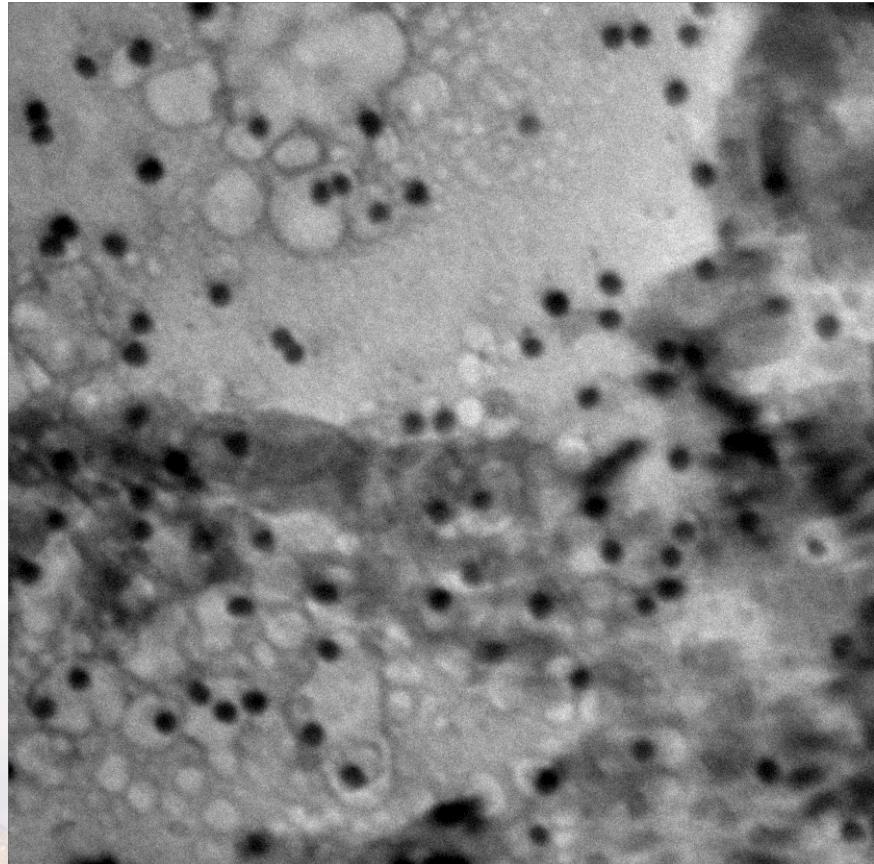
In Situ TEM observation of three-phase mixture

Collaborator: Sarah Hoppe

Au nanoparticles, aqueous solution, and gas bubbles



A pair of images before and after radiolysis from focusing the electron beam



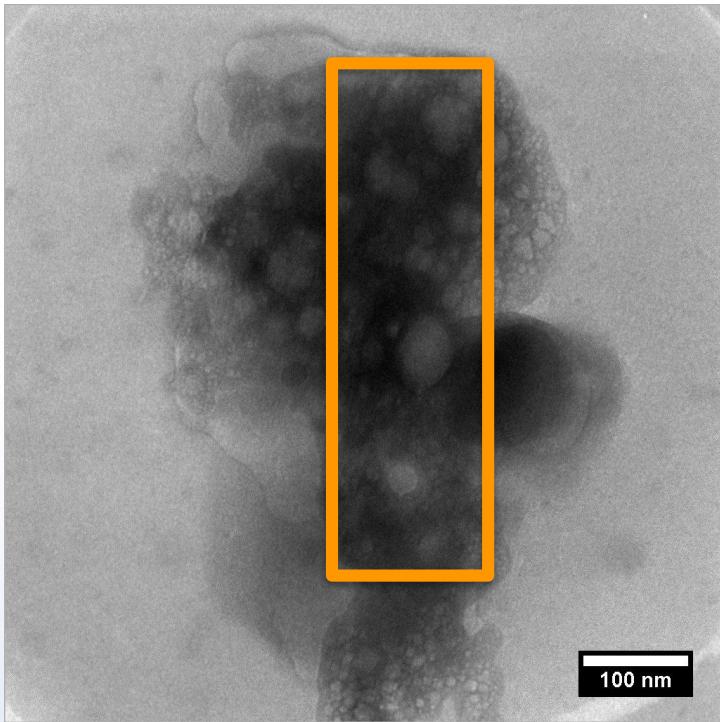
Sandia National Laboratories



PED in Liquid Cell Environment

$\text{LaCl}_3 \cdot 7\text{H}_2\text{O} : 10\text{ H}_2\text{O} : \text{PED}$

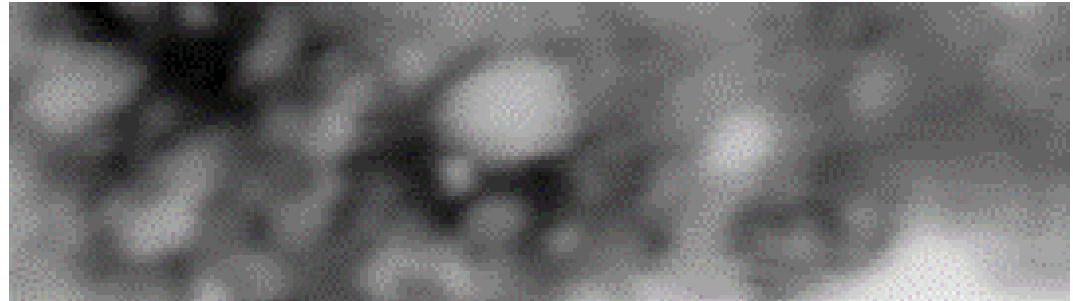
TEM Image



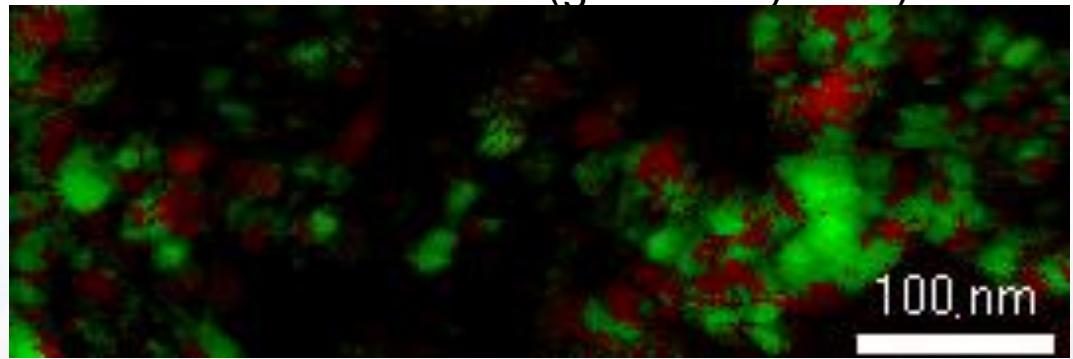
Virtual BF



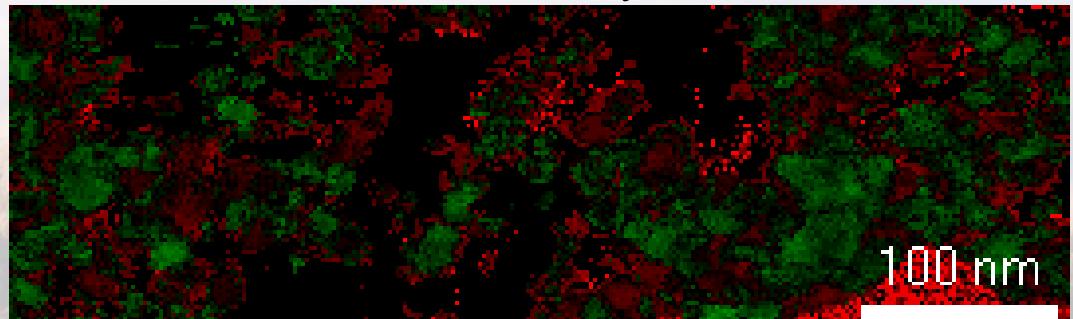
Virtual BF



Index + Phase (green = hydride)



Phase Reliability + Phase

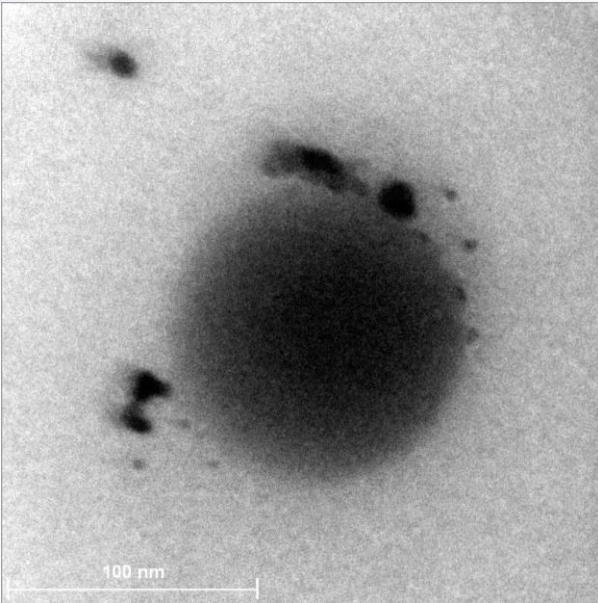


Other Fun Uses of Microfluidic Cell

Protocell Drug Delivery

S. Hoppe,
E. Carnes,
J. Brinker

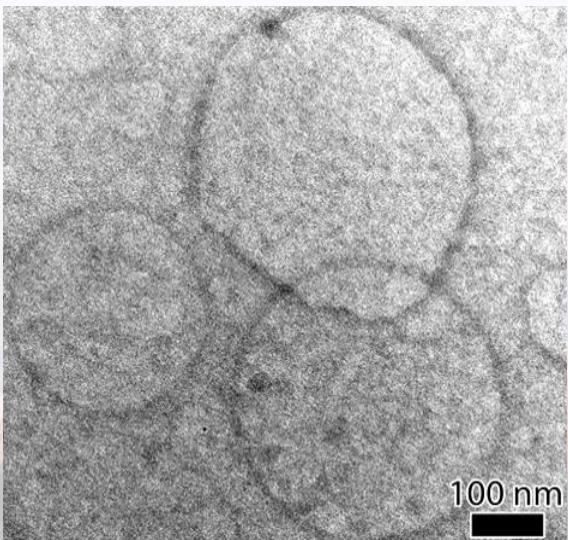
Liposome
encapsulated
Silica destroyed
by the electron
beam



Liposomes in Water

S. Hoppe,
D. Sasaki

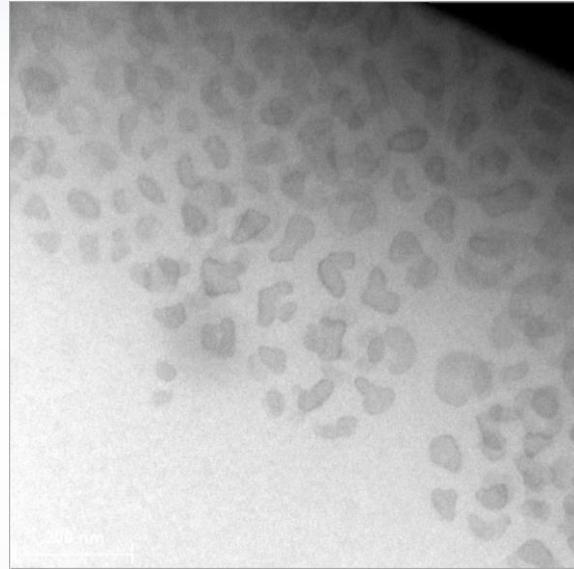
Liposomes
imaged in
flowing aqueous
channel



BSA Crystallization

S. Hoppe

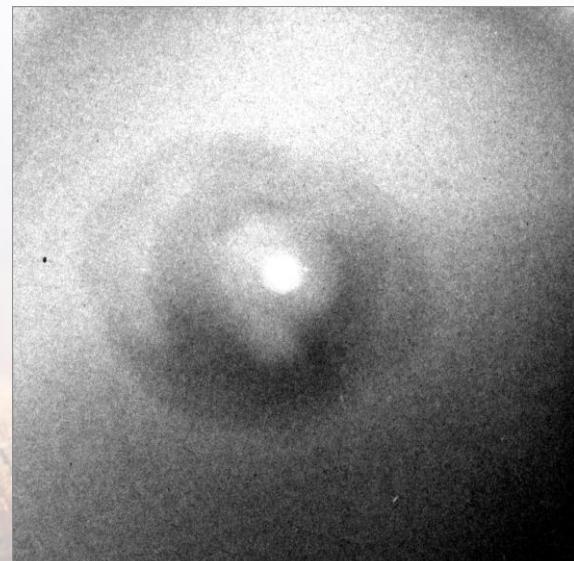
Crystallization of excess
Bovine Serum Albumen
during flow



La Structure Formation

S. Hoppe,
T. Nenoff

La
Nanostructure
form from LaCl_3
 H_2O in wet cell
due to beam
effects



Feasibility of Studying Zircaloy 2 at Nominally 1 atm

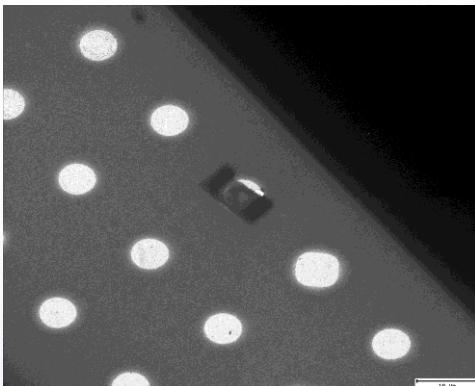
Collaborators: S. Rajasekhara and B.G. Clark



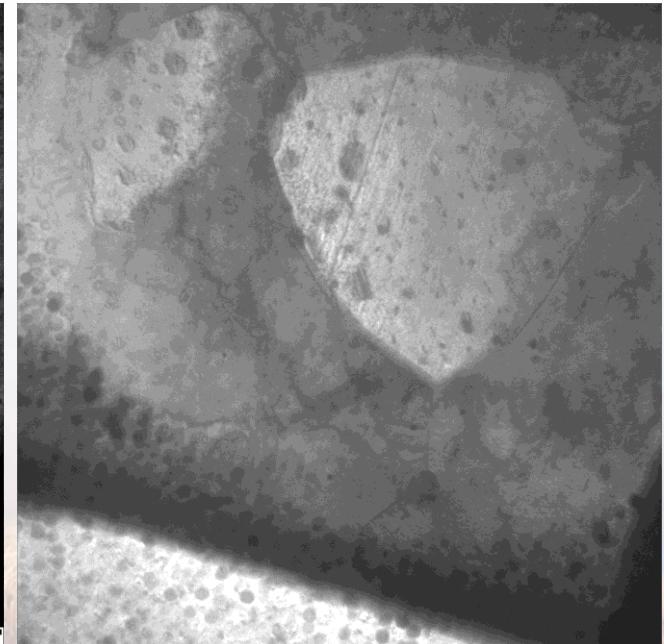
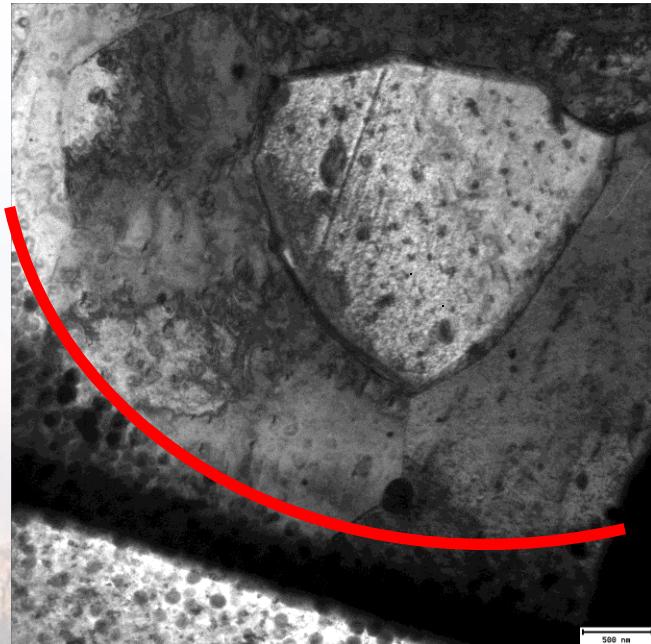
Vapor-Phase Heating TEM Stage

- Compatible with a range of gases
- *In situ* resistive heating
- Continuous observation of the reaction channel
- Chamber dimensions are controllable
- Compatible with MS and other analytical tools

Vacuum & Single Window



Nominally 1 atm H₂ & Two Windows

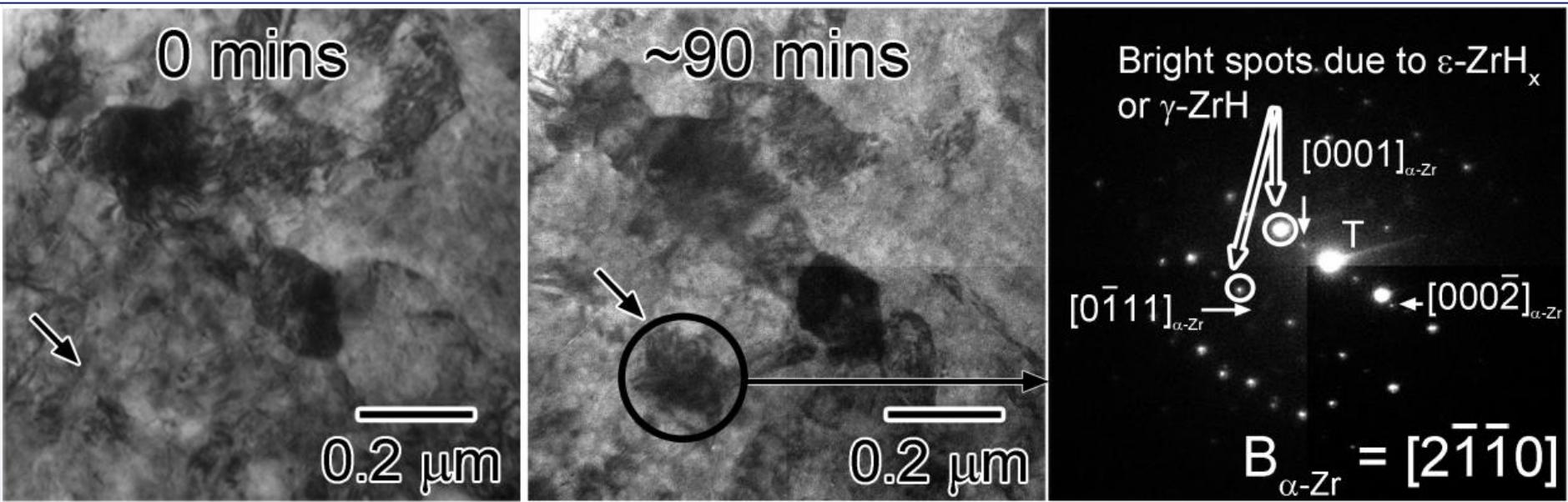


Most features are observed in both despite the decreased resolution resulting from the additional SiN window and 5 μm of air

In situ Observation of Hydride Formation in Zirlo

Collaborators: S. Rajasekhara and B.G. Clark

Absolute hydrogen pressure: 327 torr (~ 0.5 atm),
Ramp rate: 1 °C/s, Final temperature: ~ 400 C, Dwell time: ~ 90 mins

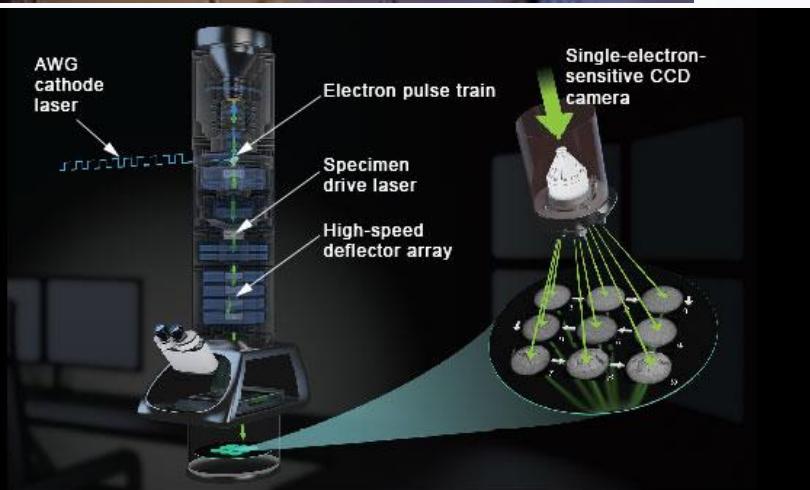
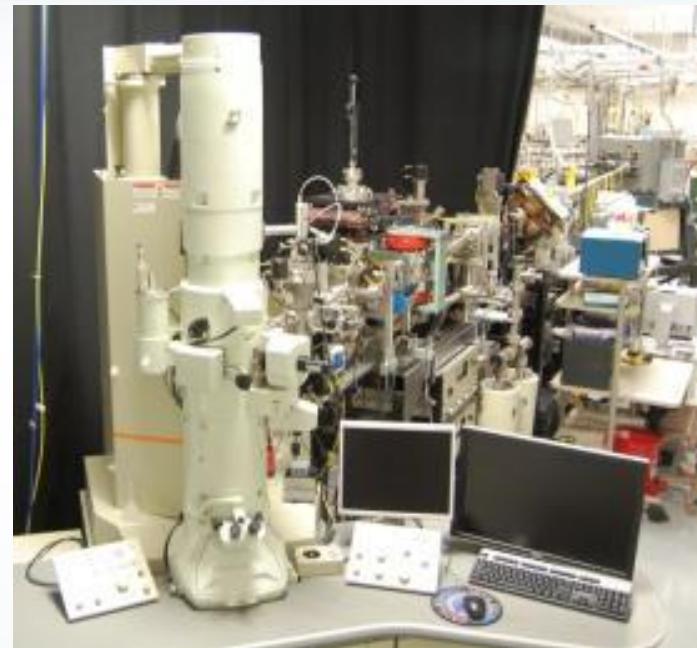


Hydride formation shown, for the first time by use of a novel TEM gas-cell stage, at elevated temperature and hydrogen pressure



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Can I³TEM and DTEM systems be combined?



Goal:

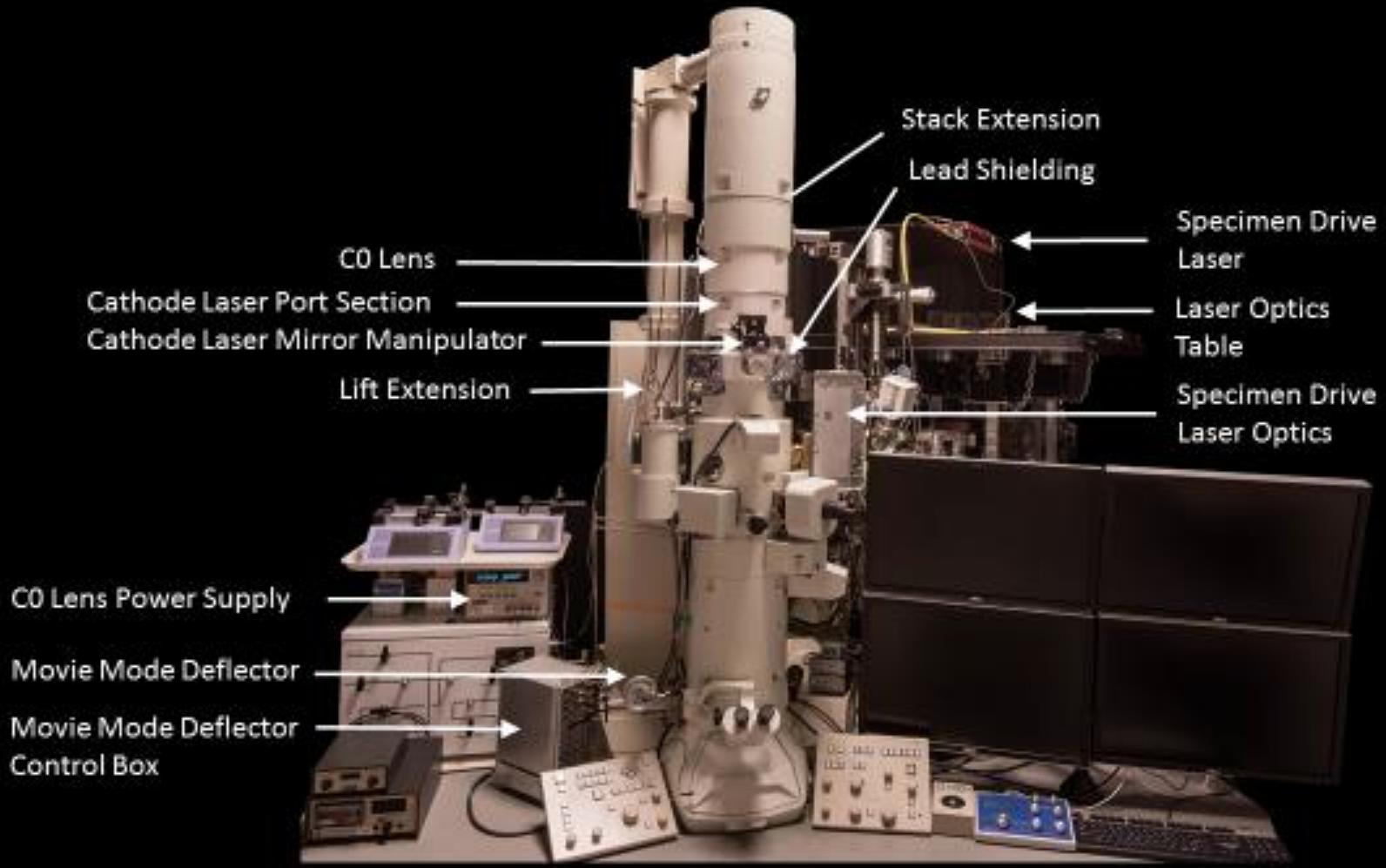
To combine the state-of-the-art in microscopy of DTEM and I³TEM to elucidate the response of extreme overlapping environments with adequate spatial and temporal resolution.



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Current Status of Laser Addition and DTEM Conversion

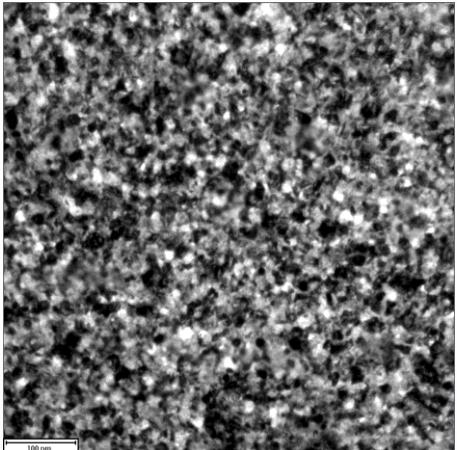


UV laser has been purchased, the safety modifications and paperwork is complete, and optics are under construction. The Ta cathode is under development. Hope for single shot observations soon!

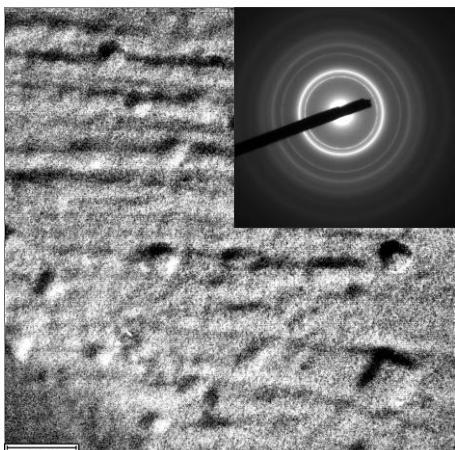
Initial Laser Heating Observations

Collaborator: C.M. Barr, D. Adams, M. Abere

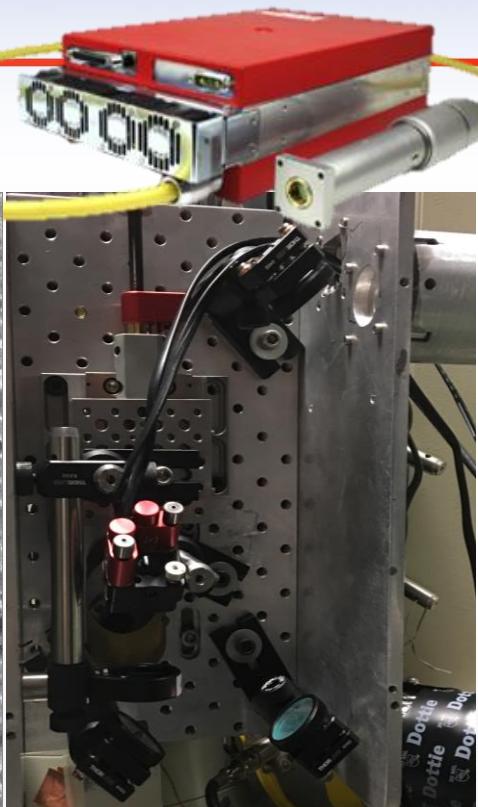
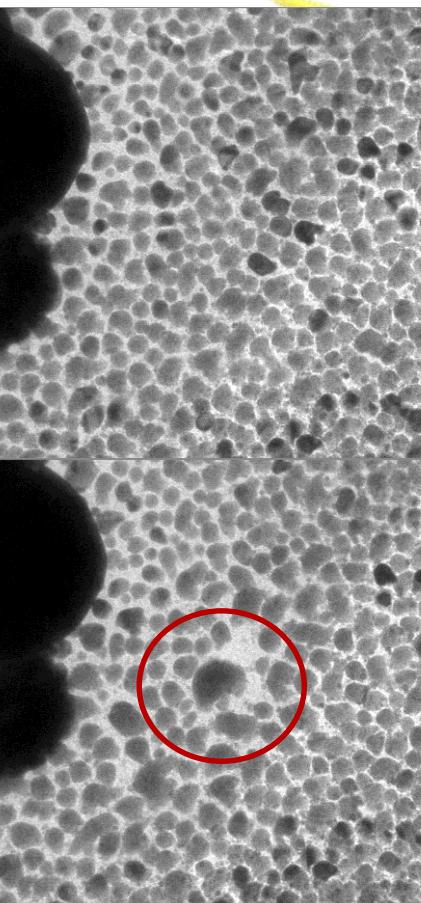
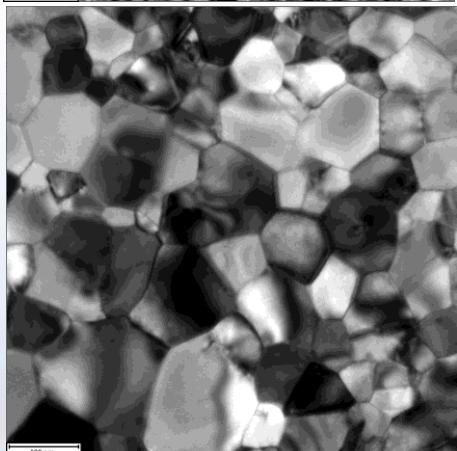
Pt Grain Growth



Reactive Multilayer Films



Nanoparticle Sintering



Electron Beam

IR Laser

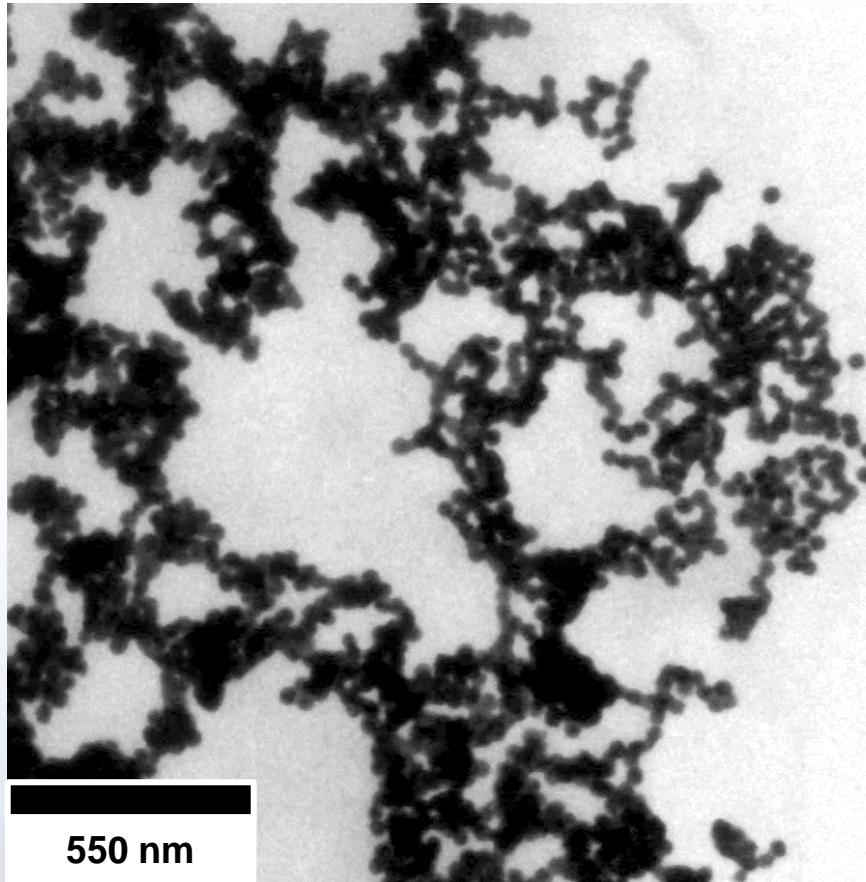
We can now introduce rapid thermal heating with
any TEM stage or ion beam conditions



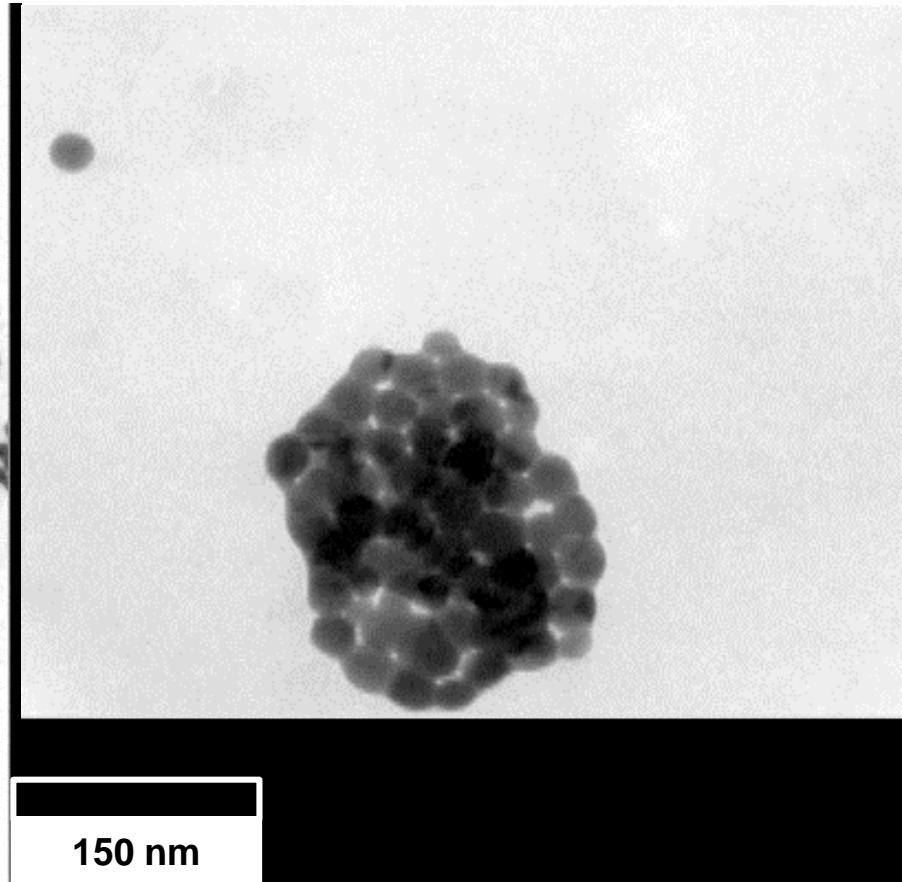
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Complex Interaction Au NPs Exposed to Laser Irradiation



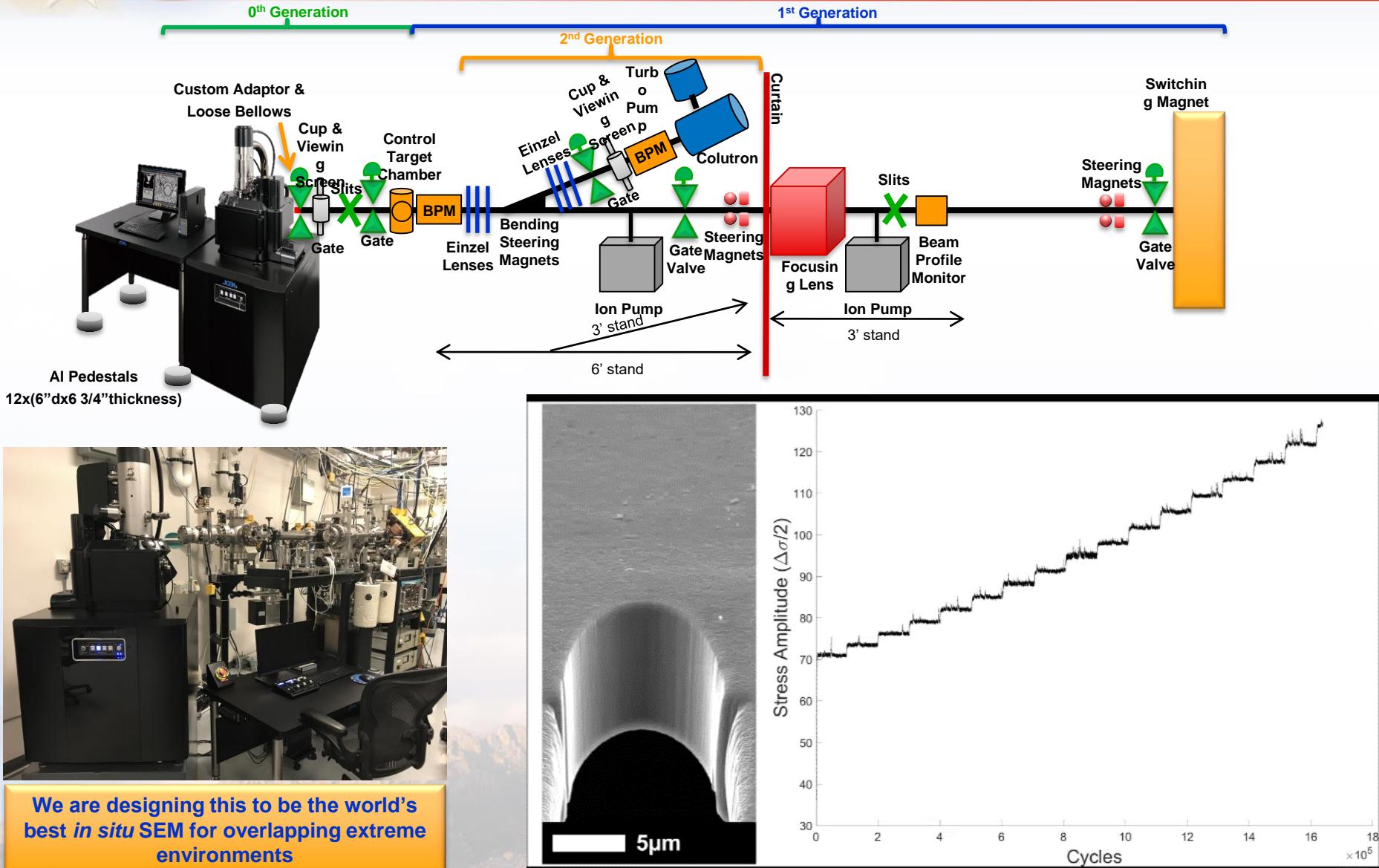
Speed = 2.5x



A Complex Combination of Sintering, Reactions, and Ablation Occurs

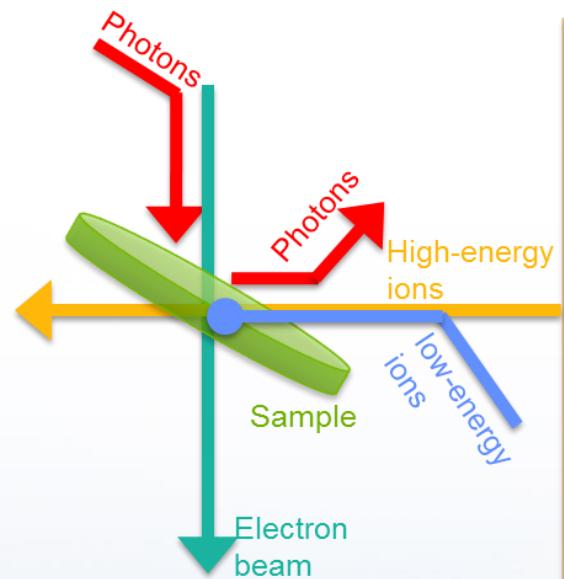
Scaling Back Up: *In situ* Ion Irradiation SEM (I³SEM)

Collaborators: N. Heckman, D. Buller, B. Boyce, J. Carroll, P. Price, C. Taylor, B. Muntifering, S. Briggs



Summary

The Ion Beam Lab at Sandia National Laboratories applies a variety of nanoscale tools to a wealth of problems

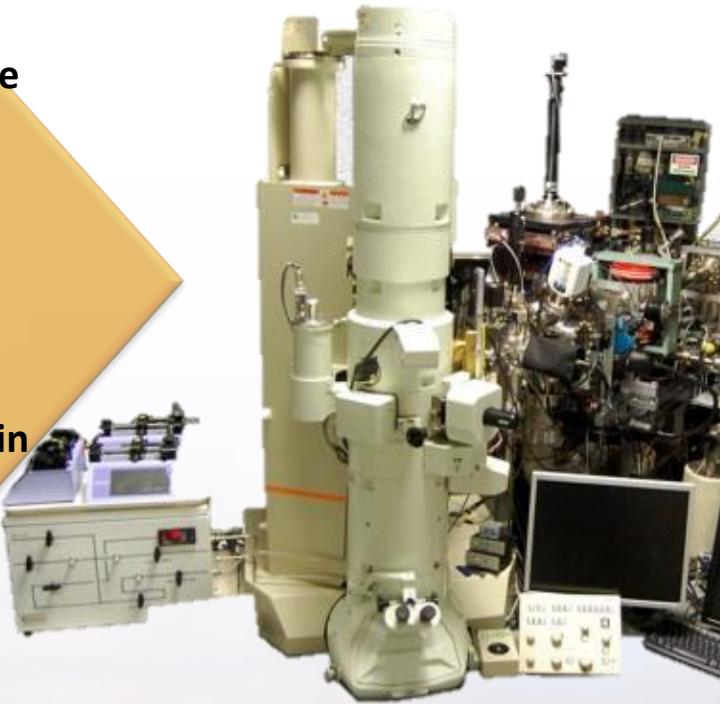


Sandia's I³TEM is one of a few in the world

- *In situ* irradiation from H to Au
- *In situ* gas implantation
- Combinations of in-situ techniques

I³TEM can provide fundamental understanding to key mechanisms in a variety of extreme conditions

The I³TEM capability are still being expanded...



Collaborators:

- IBL: **D.C. Bufford, D. Buller, C. Chisholm, B.G. Clark, B.L. Doyle, S. H. Pratt, & M.T. Marshall**
- Sandia: **B. Boyce, T.J. Boyle, P.J. Cappillino, J.A. Scott, B.W. Jacobs, M.A. Hekmaty, D.B. Robinson, E. Carnes, J. Brinker, D. Sasaki, J.A. Sharon, T. Nenoff, W.M. Mook**
- External: **A. Minor, L.R. Parent, I. Arslan, H. Bei, E.P. George, P. Hosemann, D. Gross, J. Kacher, & I.M. Robertson**

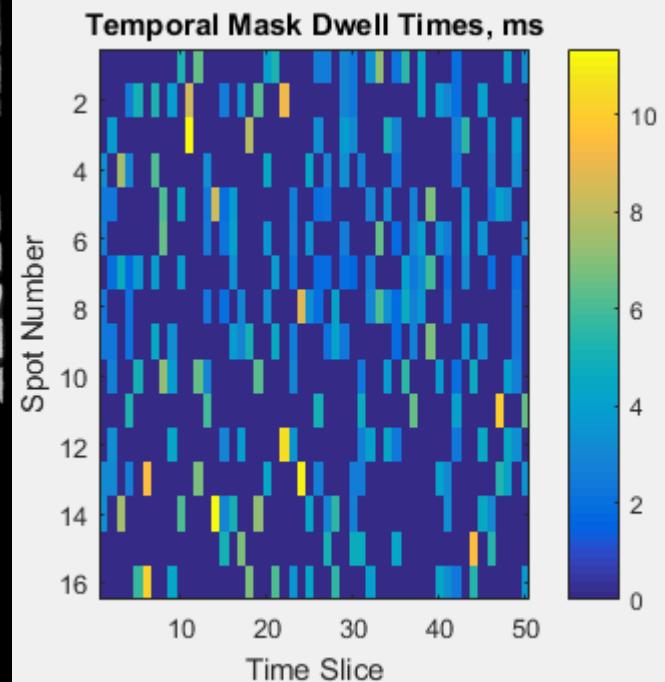
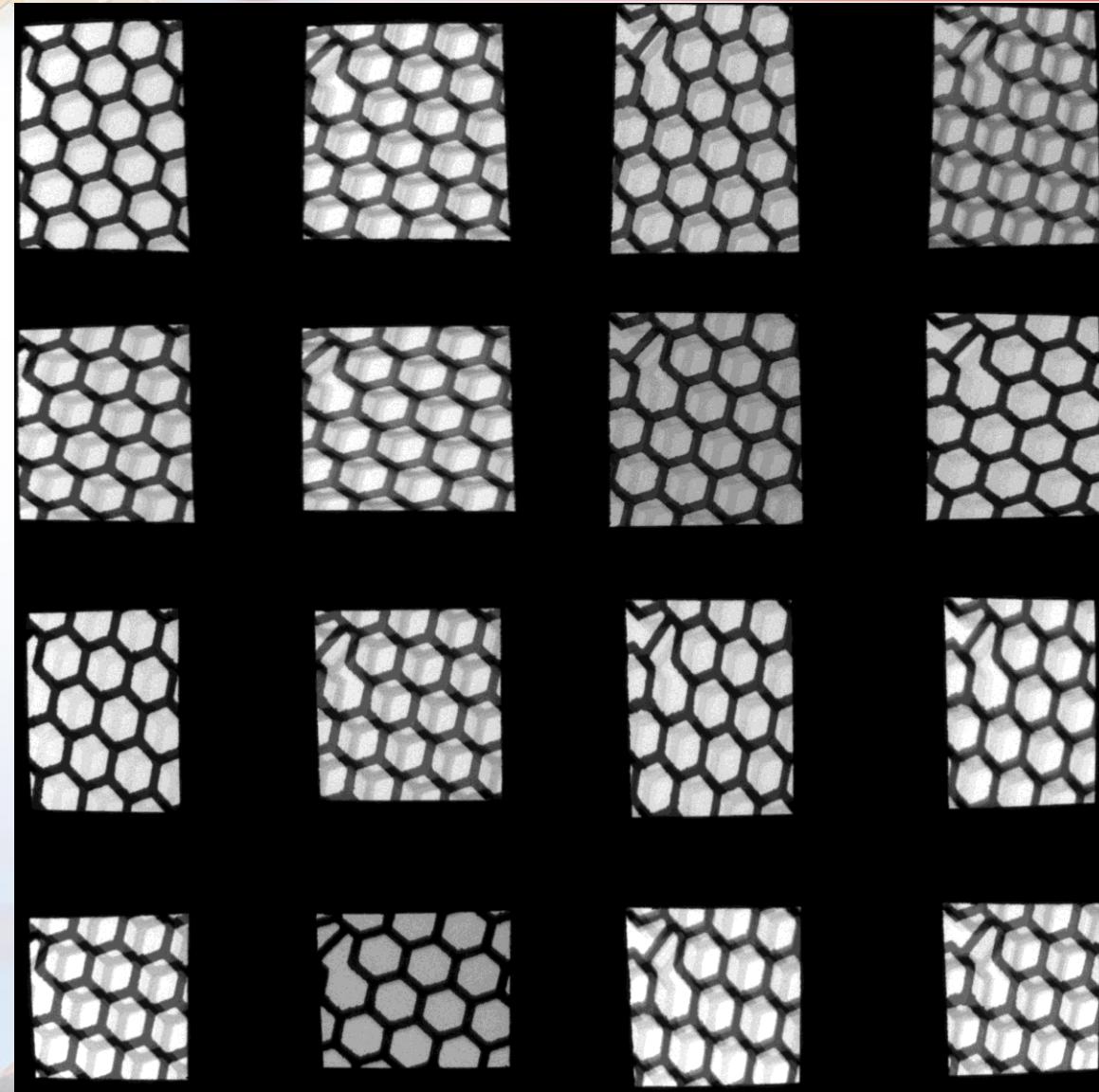


Back-up



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Demonstration of Movie Mode Deflectors



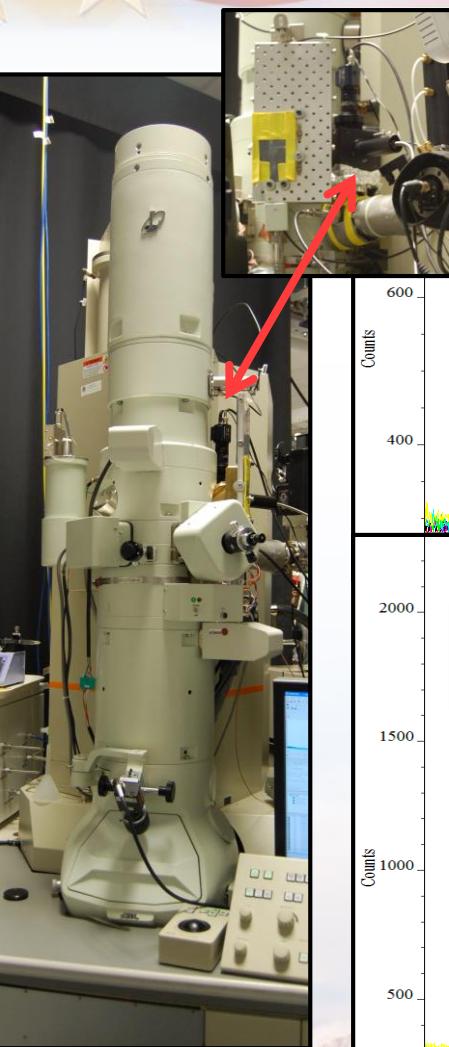
Demonstration of Relativity done
12/14/2016.



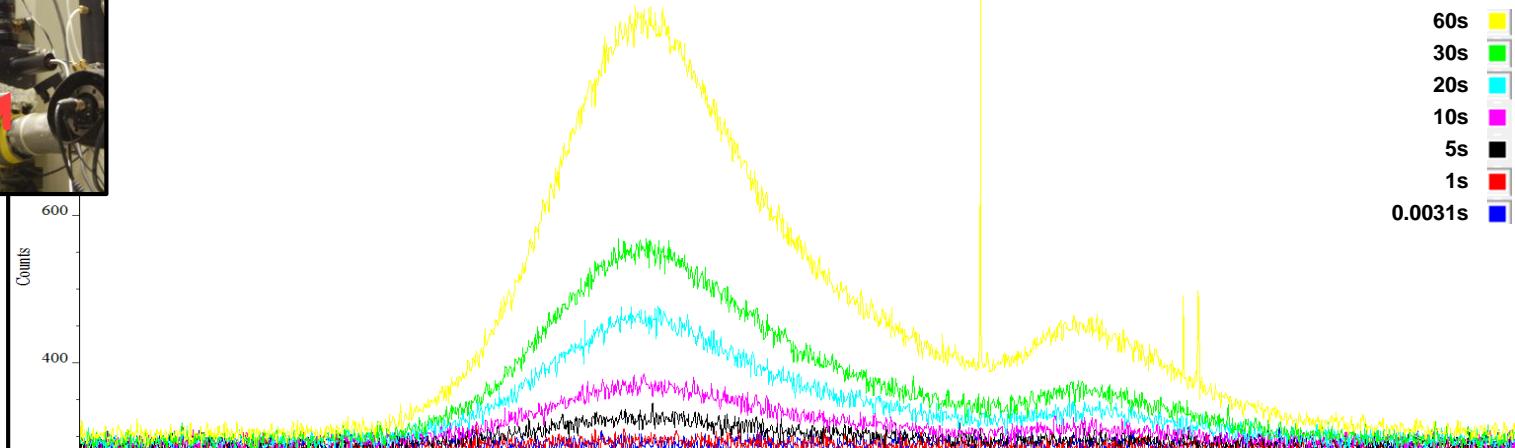
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Future Direction: *In situ* TEM Ion beam Induced Luminescence (IBIL)

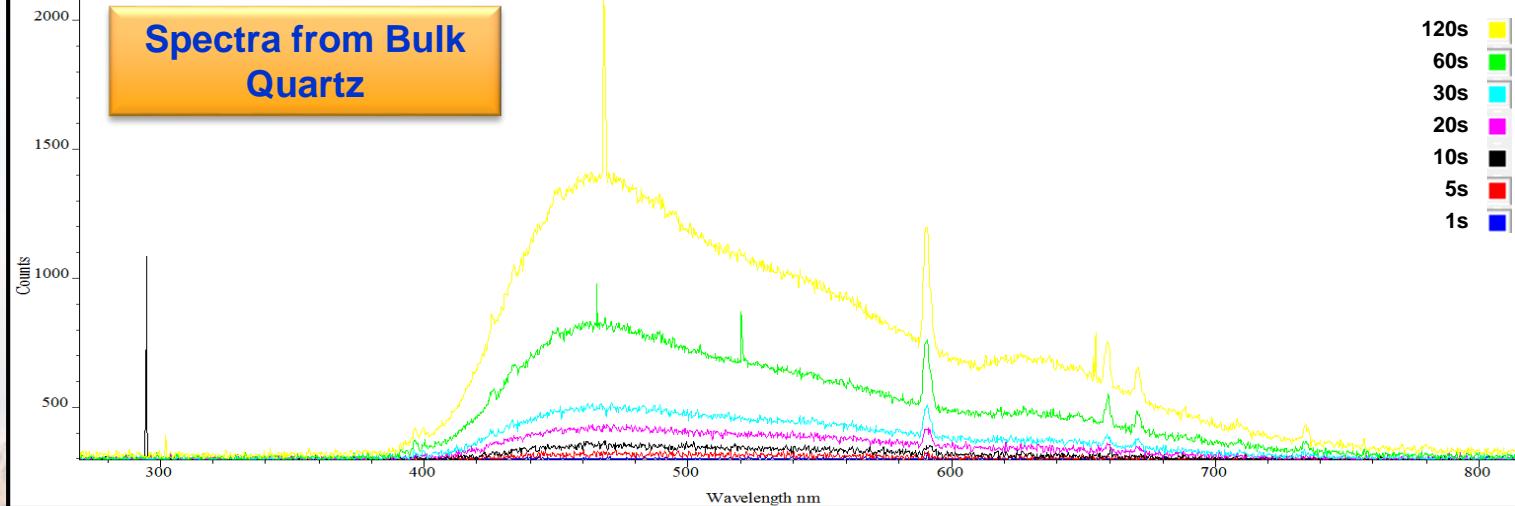
Collaborator: J. Gutierrez-Kolar



Cathodoluminescence (CL)



Ion Beam Induced Luminescence (IBIL)

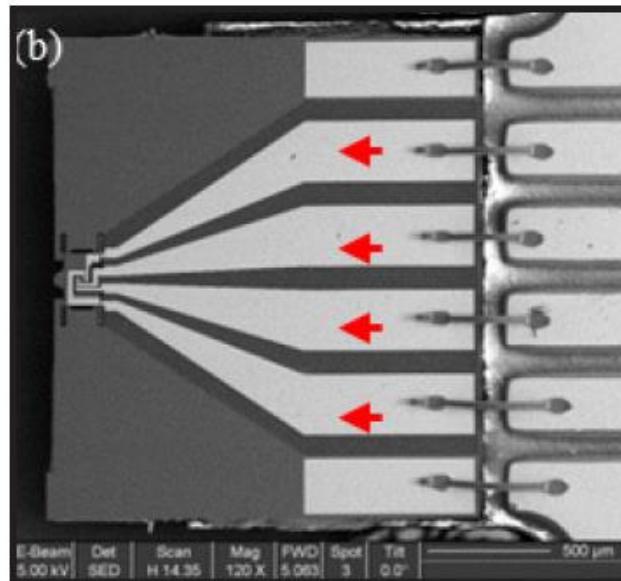
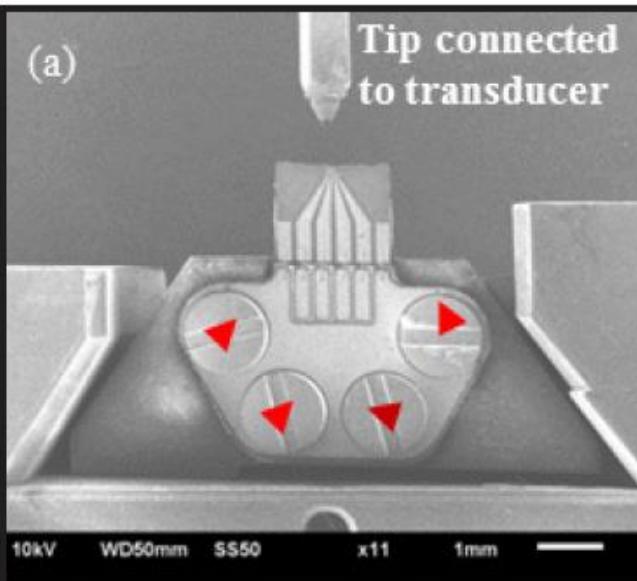


Significant optimization is still needed; potential is promising

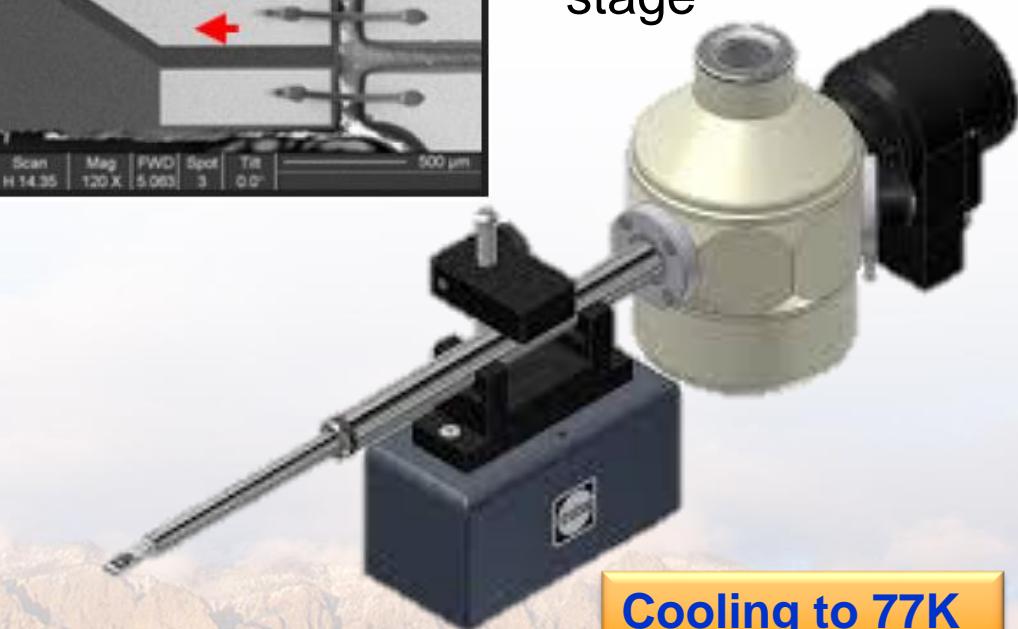
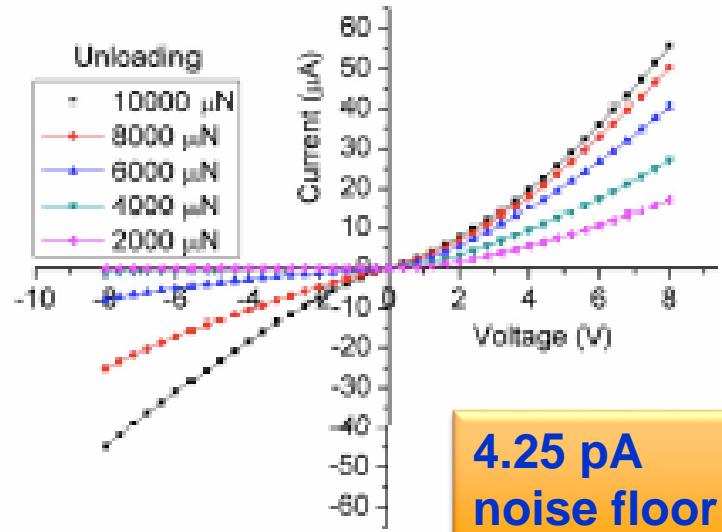


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Controlling the Sample through TEM Stage Upgrades



- Electromechanical upgrade to Hysitron PI-95
- Gatan double-tilt liquid nitrogen stage



Cooling to 77K



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