

Modeling for Metal Additive Manufacturing

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Modeling Workflow for Additive Manufacturing

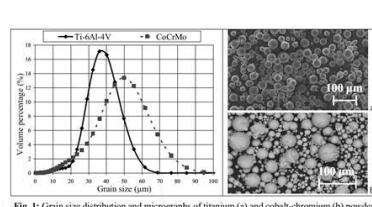
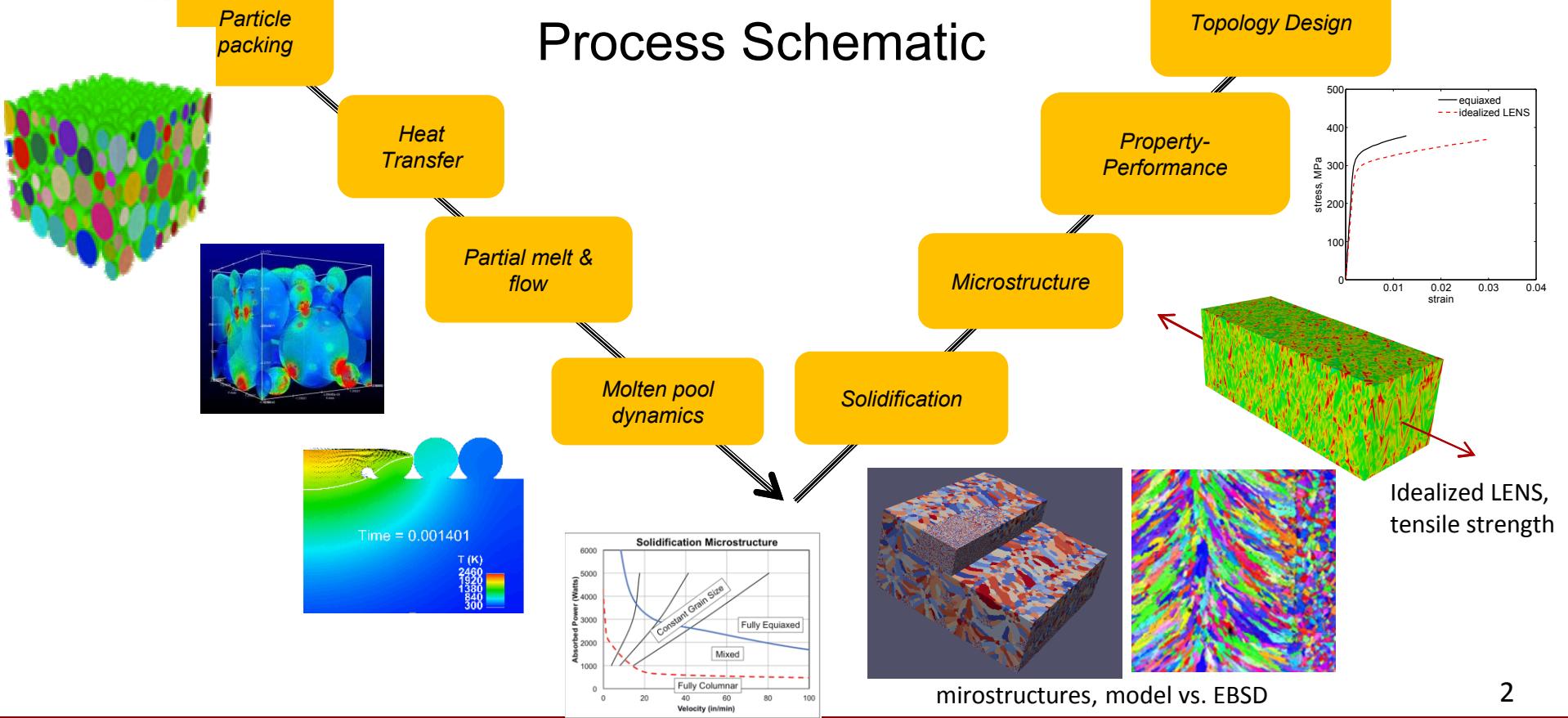


Fig. 1: Grain size distribution and micrographs of titanium (a) and cobalt-chromium (b) powders

Process Schematic

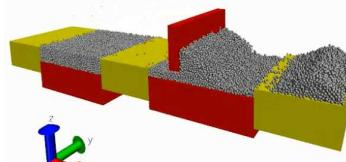


SNL Modeling Capabilities

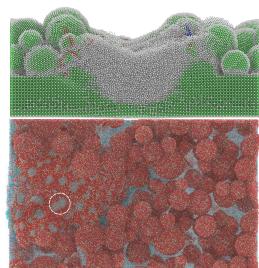
Codes

LAMMPS, SPPARKS,
Sierra/Aria,
Sierra/Adagio

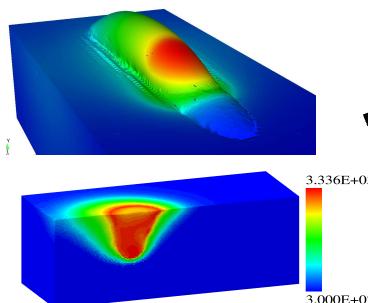
Powder Spreading
Dan Bolintineanu



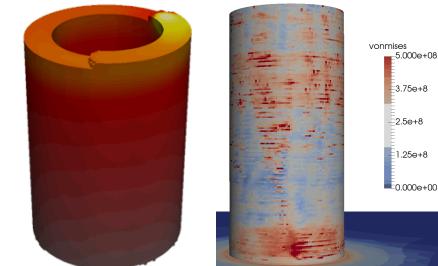
Powder Behavior
Mark Wilson



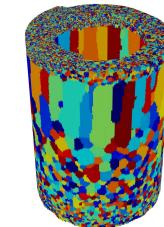
Mesoscale Selective Laser Melting
Mario Martinez & Brad Trembacki



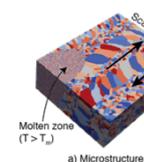
Part Scale Thermal & Solid Mechanics
Kyle Johnson, Kurtis Ford & Joe Bishop



Part Scale Microstructure
Theron Rodgers



Mesoscale Texture/Solid Mechanics/CX
Judy Brown, Theron Rodgers and Kurtis Ford



10^{-6}

10^{-3}

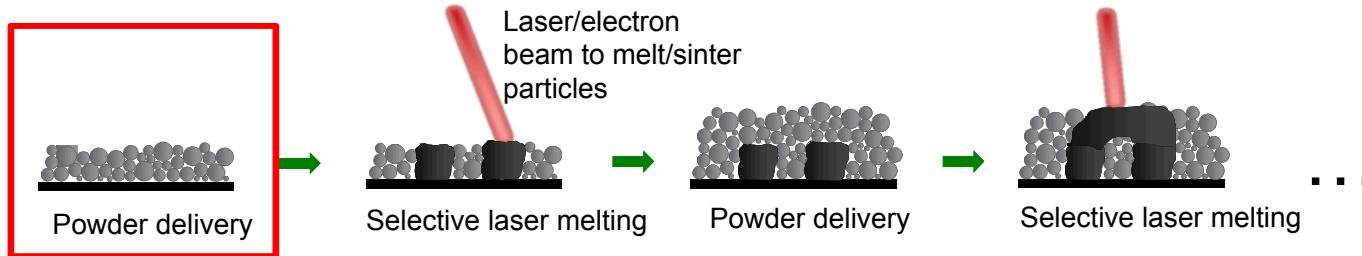
Length Scale (m)

1

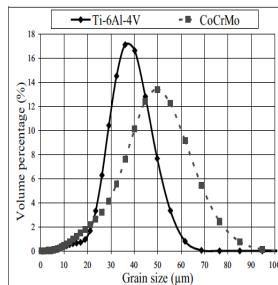
3

Powder Dynamics: Background and Motivation

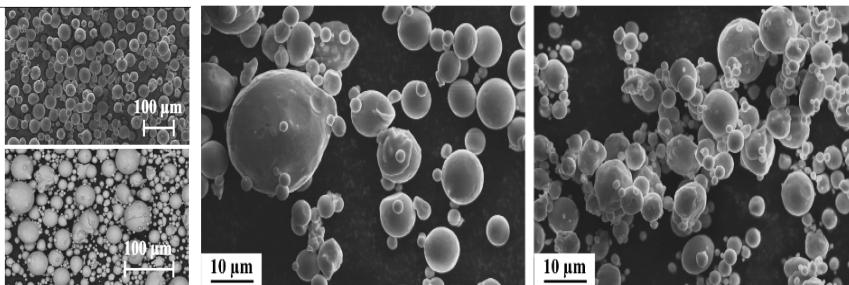
Layer-by-layer powder bed fusion processes (e.g. SLM/SLS):



- First step in AM powder bed process
 - Powder bed surface can affect laser interaction; powder bed packing can affect void formation, surface finish, thermal properties
 - Informs downstream process models
- Variability in powder properties due to vendor supply, powder recycling
- Several key process length scales are comparable to individual particles:



From Ref. 1



From Ref. 2

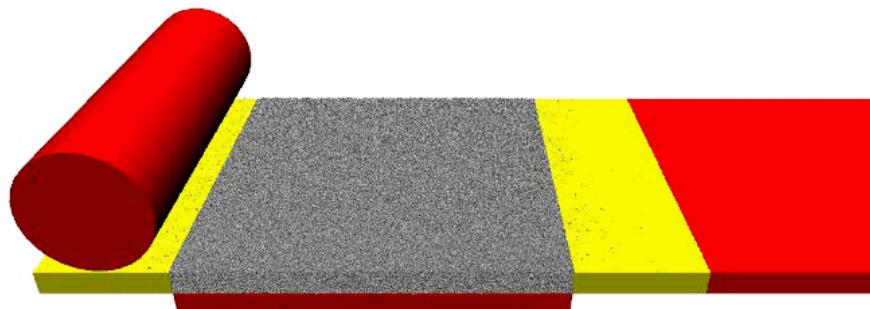
Typical particle diameter: 10-100 µm

Powder layer thickness 30-150 µm
Laser beam spot size 70-200 µm (ref. 1)

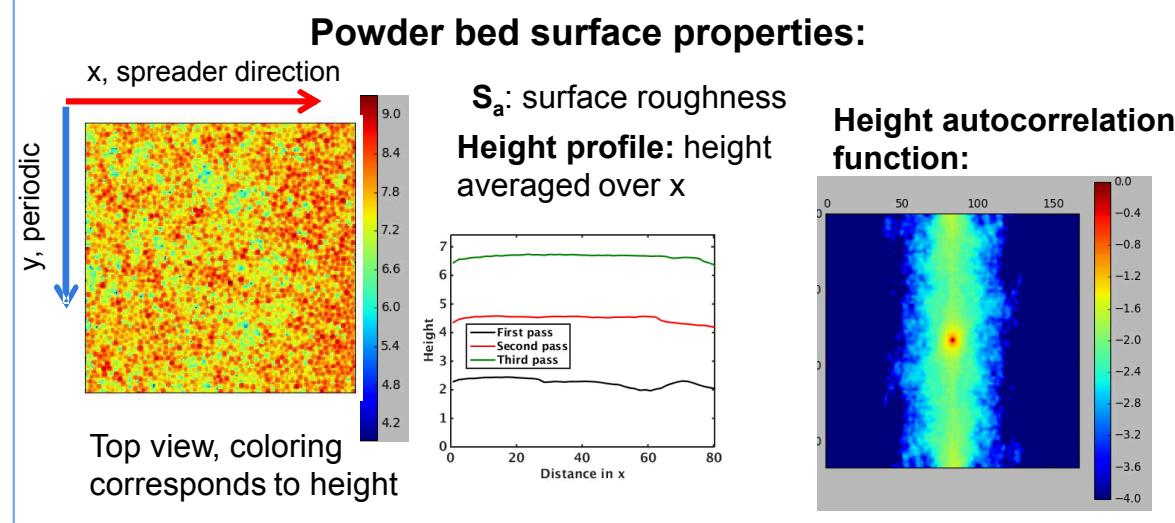
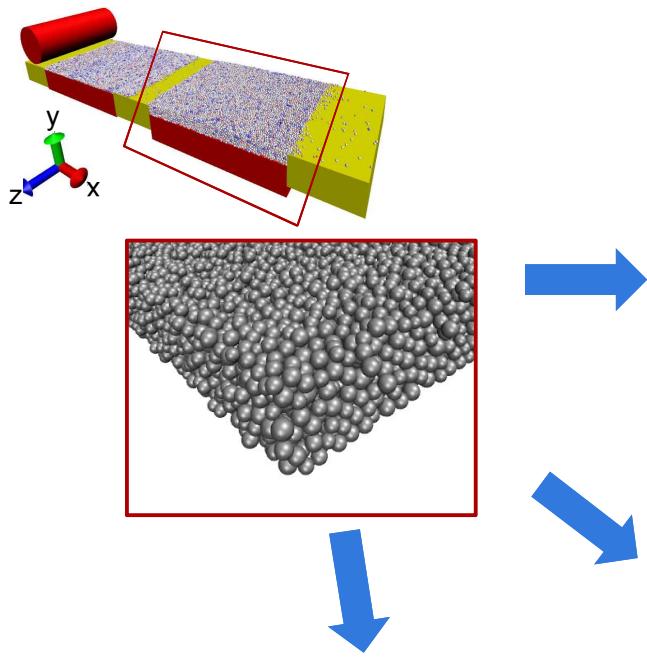
1. Vandenbroucke, B. and Kruth, J.P. *Rapid Prototyping Journal* 13 (2007): 196
2. Yadroitsev, I., et al. *Journal of Laser Applications* 25 (2013): 052003

Modeling powder dynamics using DEM

- **Discrete Element Modeling:** molecular dynamics-like method
 - Each particle modeled explicitly (position, velocity, angular velocity)
 - Forces/torques computed at contact using reduced order models
 - Dynamics integrated in time for large collection of interacting particles

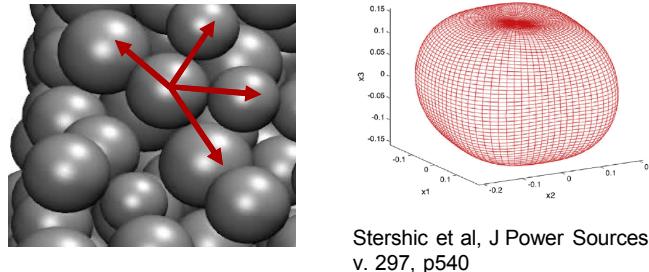


Large resulting data:



Local particle contact properties

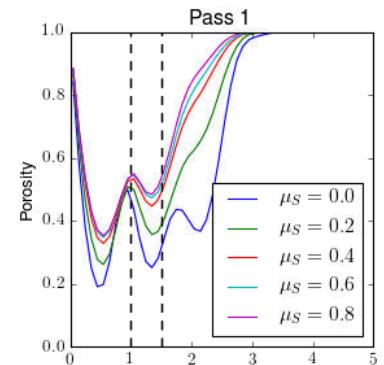
- Coordination number distribution
- Contact size distribution
- Fabric tensor



Powder bed bulk properties:

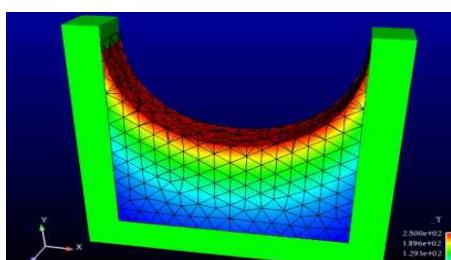
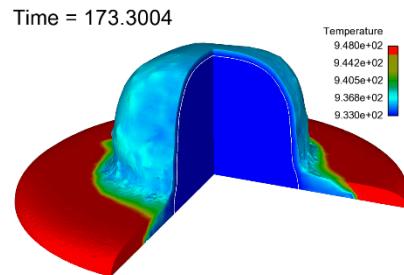
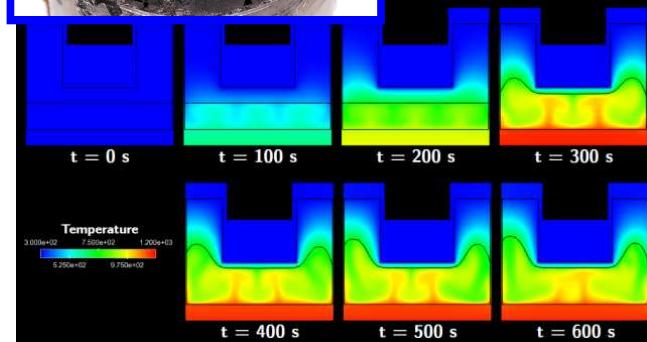
- Porosity(x), porosity(z)
- Pore size distribution
- Two-point correlation function
- 'Coarseness': variability as a function of length scale
- Lineal path function, chord length distribution

...

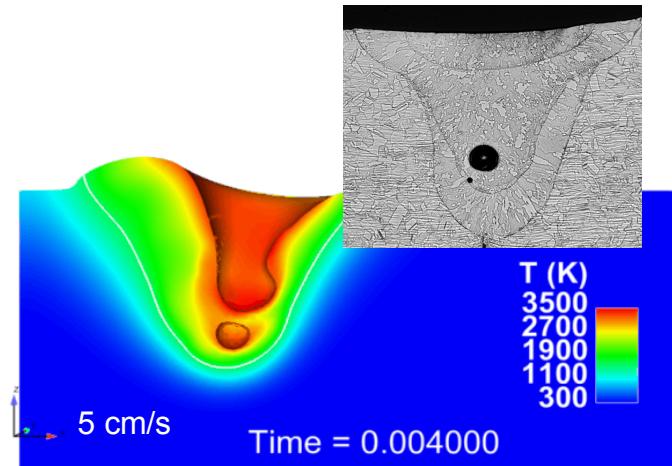


Relevant Sandia Thermal/Fluid Applications to Burn and Melt using Enriched FEM

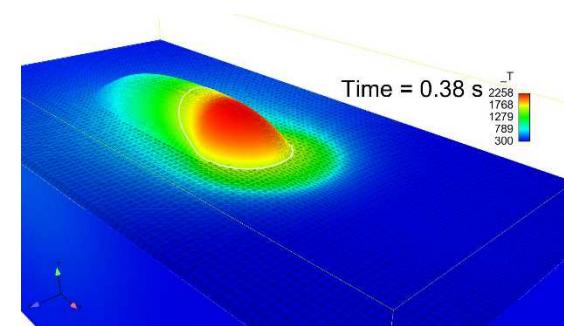
Numerous problems with moving or topologically complex interfaces with discontinuous physics and fields



Material death



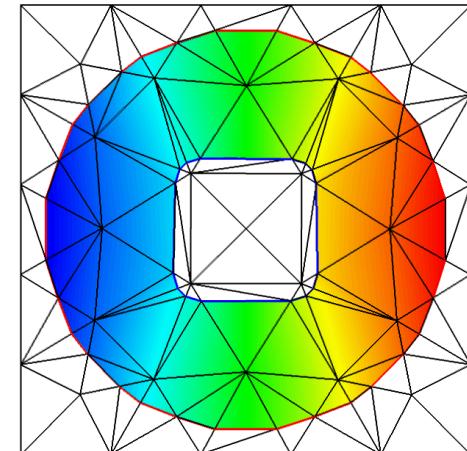
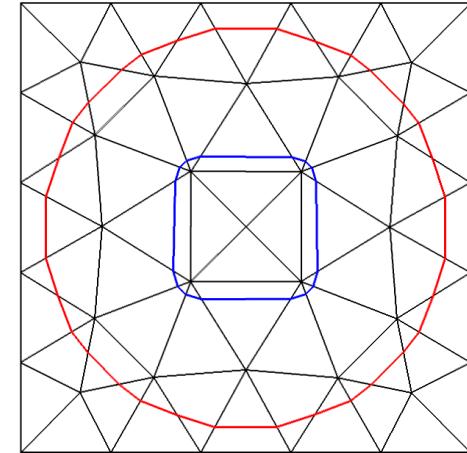
Laser welding



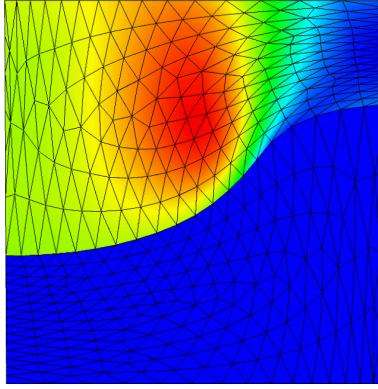
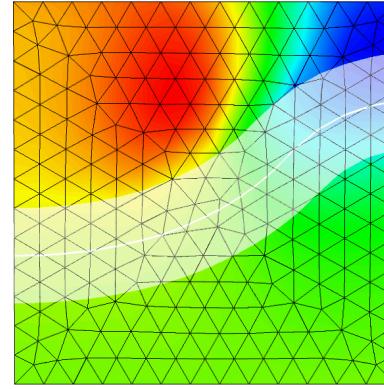
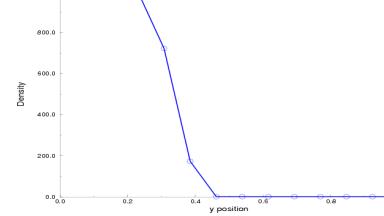
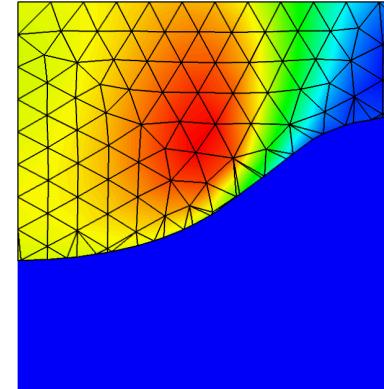
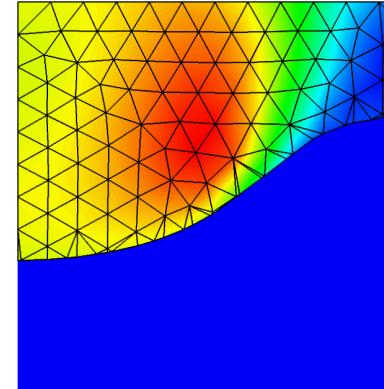
Additive Manufacturing

Conformal Decomposition Finite Element Method (CDFEM)

- Simple Concept (Noble, et al. 2010)
 - Use one or more level set fields to define materials or phases
 - Decompose non-conformal elements into conformal ones
 - Obtain solutions on conformal elements
- Related Work
 - Li et al. (2003) FEM on Cartesian Grid with Added Nodes
 - Ilinca and Hetu (2010) Finite Element Immersed Boundary
 - S. Soghrati and P.H. Geubelle (2012) Interface Enriched Finite Element
- Properties
 - Supports wide variety of interfacial conditions (identical to boundary fitted mesh)
 - Avoids manual generation of boundary fitted mesh
 - Supports general topological evolution (subject to mesh resolution)
- Similar to finite element adaptivity
 - Uses standard finite element assembly including data structures, interpolation, quadrature



Finite Element Methods for Moving Interfaces in Fluid/Thermal Applications Tested at Sandia

Enriched Finite Element Methods			
ALE	Diffuse LS	XFEM	CDFEM
<ul style="list-style-type: none">• Separate, static blocks for gas and liquid phases• Static discretization 	<ul style="list-style-type: none">• Single block with smooth transition between gas and liquid phases• Static discretization  	<ul style="list-style-type: none">• Single block with sharply enriched elements (weak or strong) spanning gas and liquid phases• Interfacial elements are dynamically enriched to describe phases 	<ul style="list-style-type: none">• Separate, dynamic blocks for gas and liquid phases• Interfacial elements are dynamically decomposed into elements that conform to phases 

Formulation: Interface Dynamics

- Level Set Equation

- Advection equation

$$\frac{\partial \phi}{\partial t} + \mathbf{u} \cdot \nabla \phi = 0$$

- Galerkin, Backward Euler

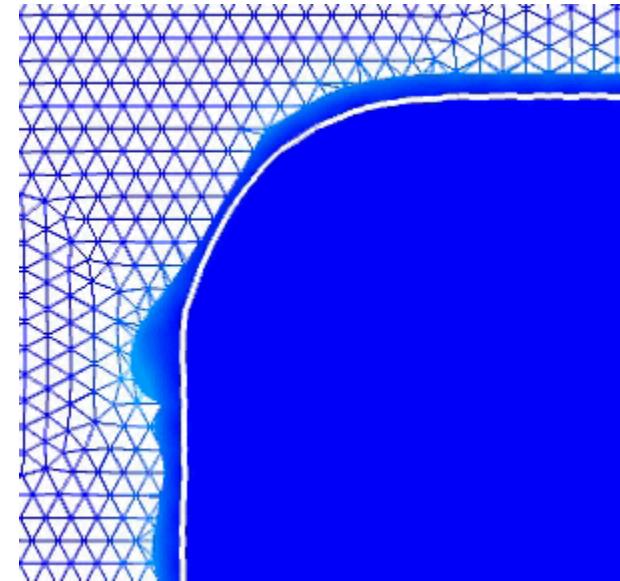
$$\int_{\Omega} \frac{\phi - \phi^n}{\Delta t} N_i \, d\Omega + \int_{\Omega} \mathbf{u} \cdot \nabla \phi N_i \, d\Omega = 0$$

- SUPG stabilization

$$N_i \Rightarrow N_i + \tau_{\phi} \mathbf{u} \cdot \nabla N_i, \quad \tau_{\phi} = \left[\left(\frac{2}{\Delta t} \right)^2 + u_i g_{ij} u_j \right]^{-\frac{1}{2}}$$

- Periodic renormalization

- Compute nearest distance to interface



Models: Liquid-Air Interface

■ Capillary Force

- Same model used in ALE simulations
 - Jump in stress due to interfacial tension

$$\int_{\Gamma} (\gamma \kappa \mathbf{n} + \nabla_s \gamma) N_i \, d\Gamma = \int_{\Gamma} \gamma \nabla_s N_i \, d\Gamma, \quad \nabla_s \equiv (\mathbf{I} - \mathbf{n}\mathbf{n})\nabla$$

■ Interface Stabilization

- Surface viscosity type stabilization
 - Based on recent paper by Hysing

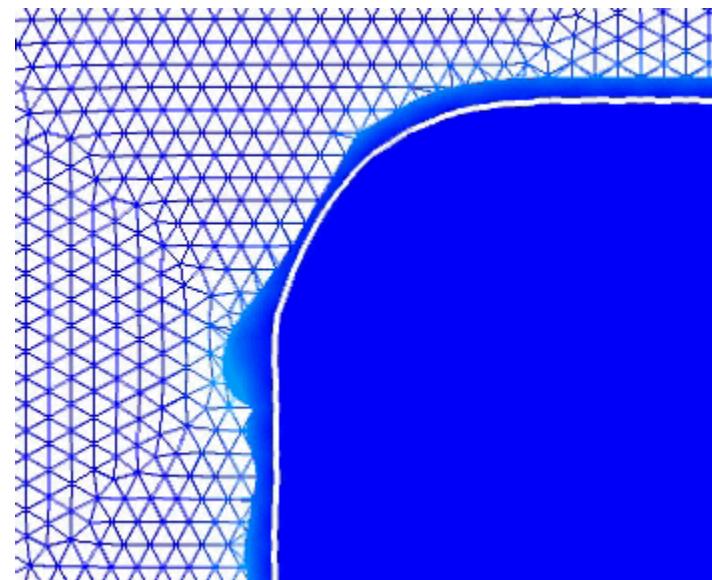
$$\int_{\Gamma} \mu_s \nabla_s u \cdot \nabla N_i \, d\Gamma$$

■ Radiation

- Simple radiation boundary condition

$$\int_{\Gamma} \varepsilon \sigma (T^4 - T_e^4) N_i \, d\Gamma$$

- Enclosure radiation
 - Enclosure temperature 2000K
 - Repeat viewfactor calculation every time step

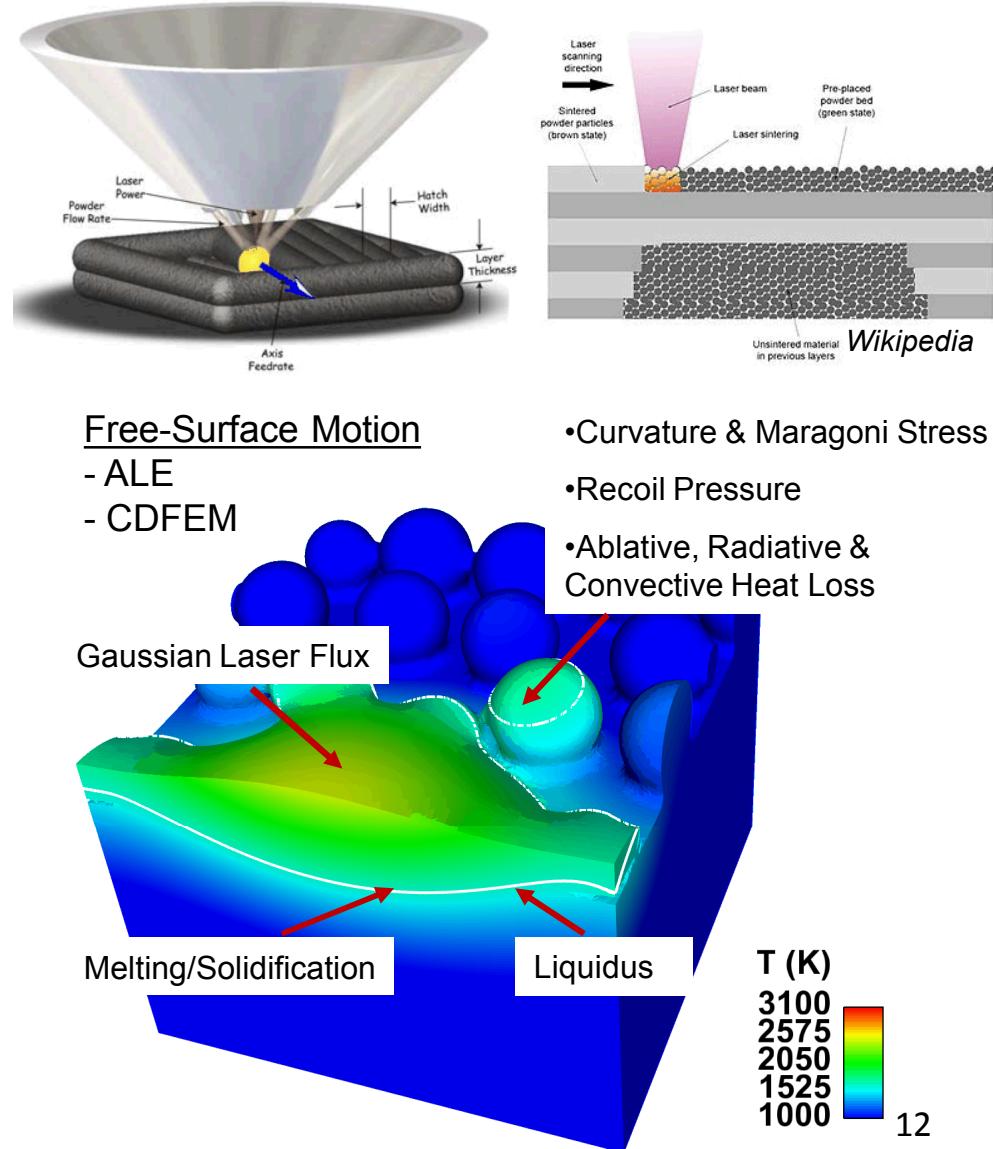


Powder Bed Physics Model

Goal: Link AM mesoscale processes to macroscale performance

Method: Conformal level-set technology includes melt and ambient (or assist) gas dynamics

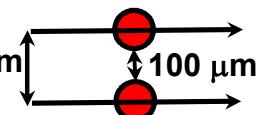
- Laser energy coupling to particle packs
- Melt/solidification, capillary-driven flow, buoyant gas convection, solutal segregation
- Impact of laser setting: power, spot size, scan rate, hatch spacing, ...
- Laser schedule: edge modulation, variable power, variable spot,
- Beam overlap, remelt, porosity



SNL Prox 300 Model

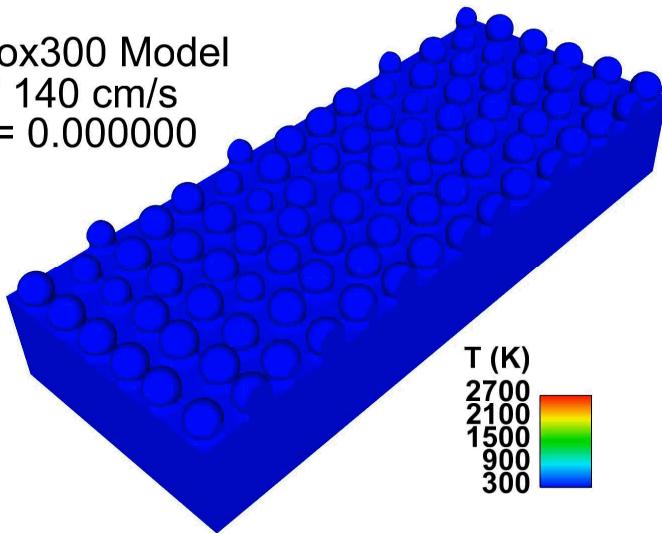
Impact of power for fixed hatch spacing

Hatch Spacing: 150 μm



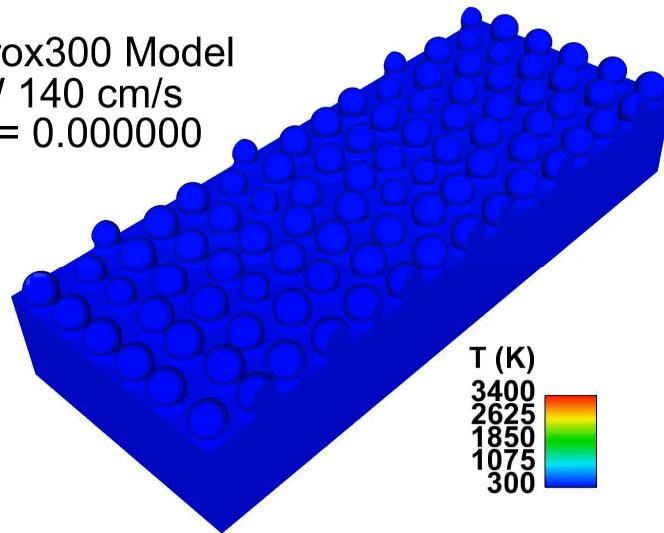
100 μm

SNL Prox300 Model
25W 140 cm/s
Time = 0.000000



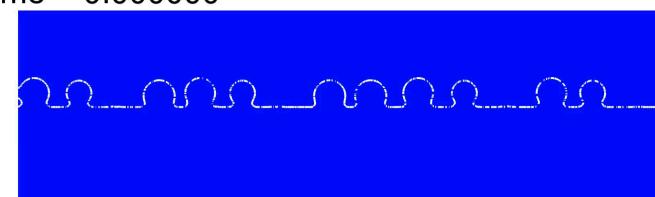
Stainless steel 304L
25 micron powder

SNL Prox300 Model
50W 140 cm/s
Time = 0.000000



Gas and melt pool dynamics

Time = 0.000000



Notes:

- 500 micron powder bed traversed in **357 microsec.**
- Sloshing-driven **gas dynamics** entrains ambient gas

T (K)

3400
2625
1850
1075
300

SNL Prox 300 Model

Gaussian Laser 140 cm/s

General features:

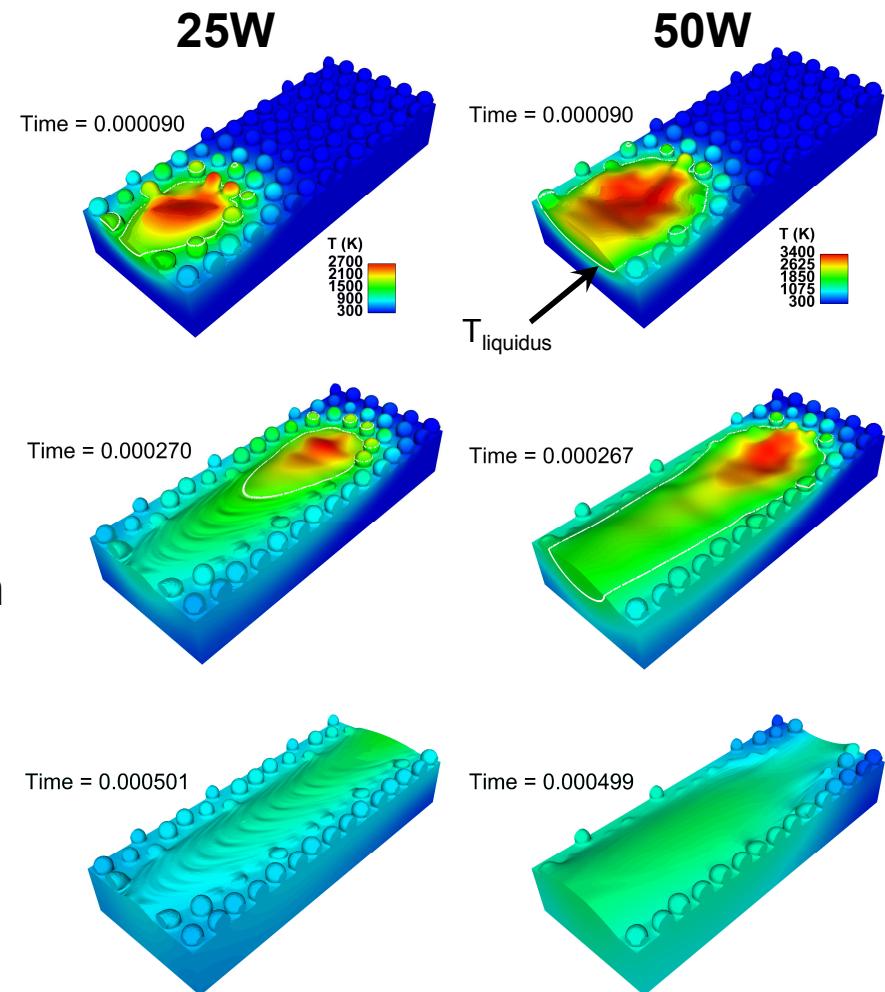
- Surface flux melts particles top to bottom;
- Fully molten particles jump into melt pool by capillarity, inciting sloshing

25W

- Shallow melt pool; smaller melt path
- Solidification front freezes in wave peaks and troughs – *ribbed finish*

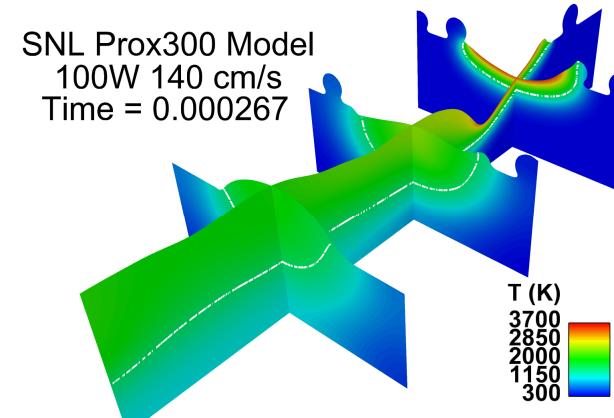
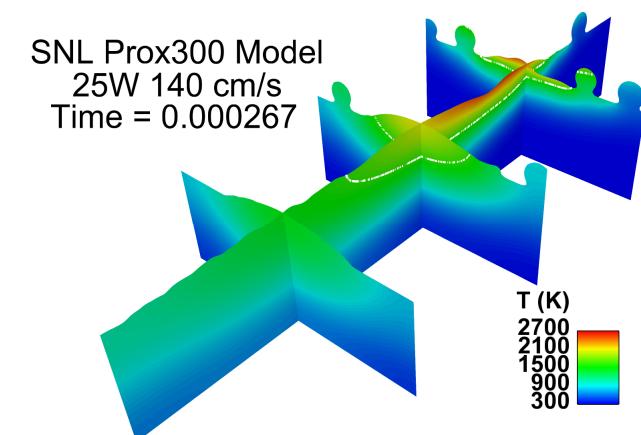
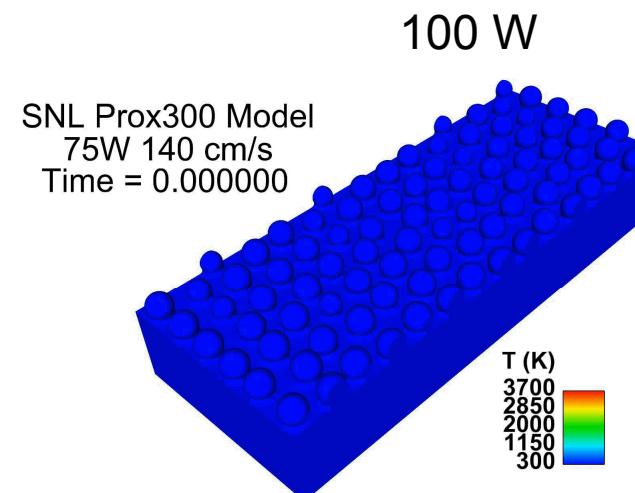
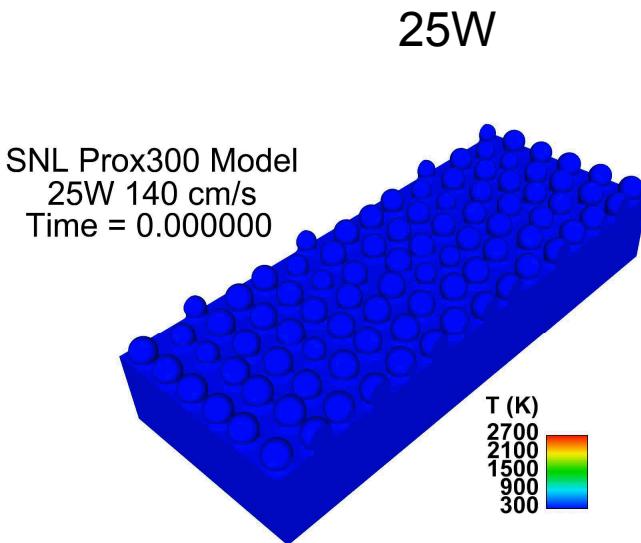
50W

- Long, deep melt pool; wider melt path
- *Smooth finish* (?)
- Laser power modulated at exit



Power, hatch spacing, and vertical heat penetration can be guided/optimized with melt dynamics models.

Power and recoil pressure



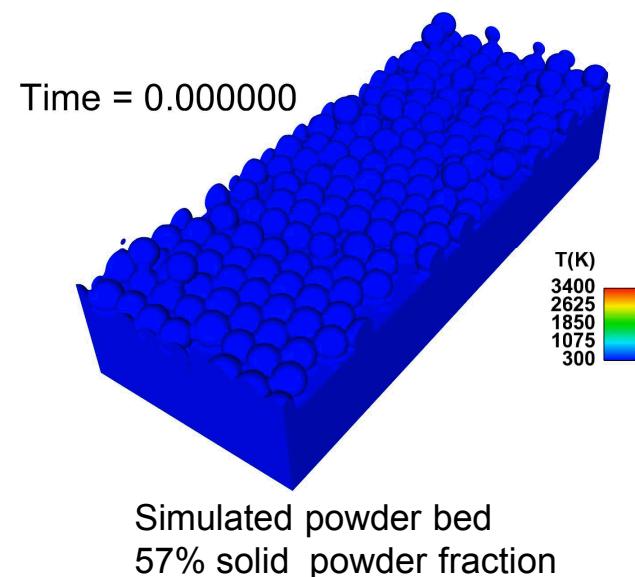
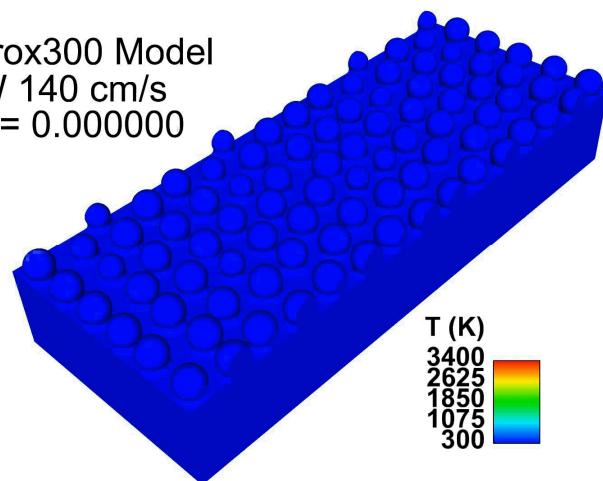
- Small melt pool
- Freezes in surface ripples

- Keyhole (small) at high power
- Recoil pressure → balling effect

Powder porosity

50W $R_{eff}=0.6$

SNL Prox300 Model
50W 140 cm/s
Time = 0.000000



Lower porosity powder:

- requires more laser power
- Higher void probability

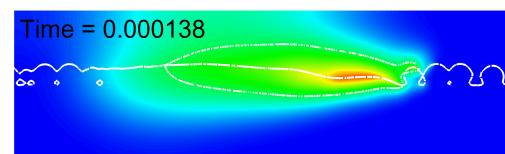
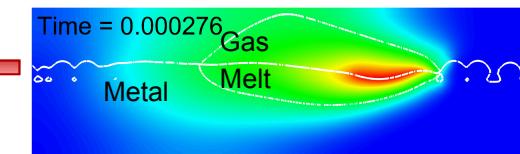
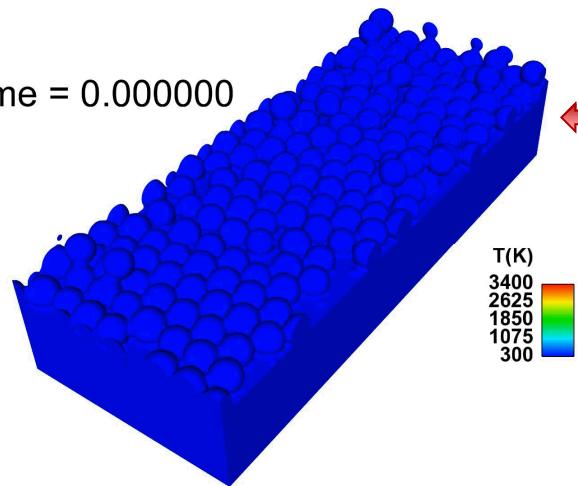
Laser Scan Rate

Energy deposition rate $\sim Q/v$

50W Reff=0.6

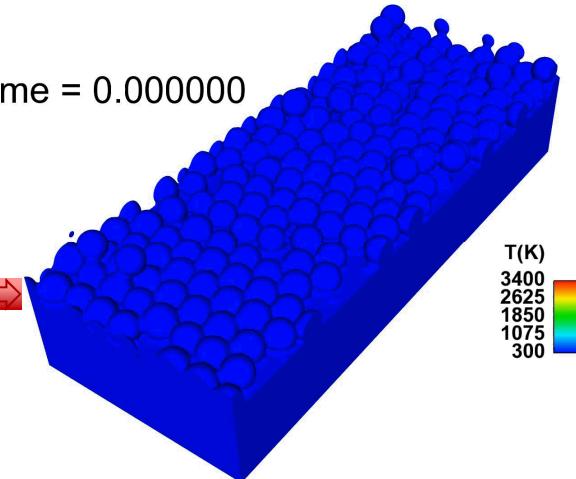
100 cm/s

Time = 0.000000

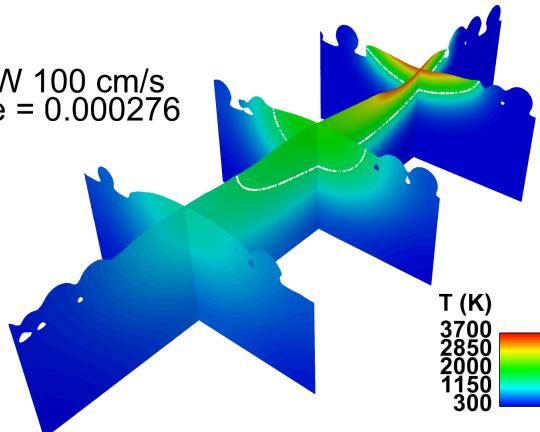


200 cm/s

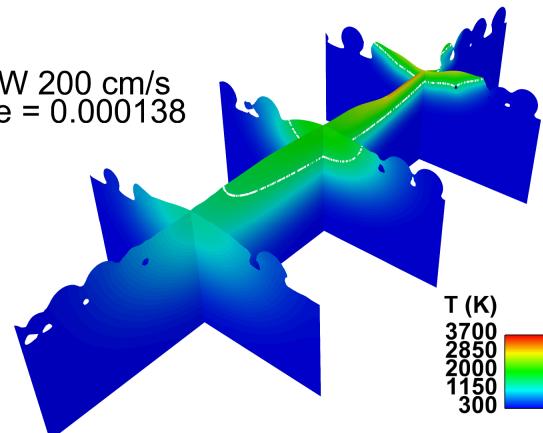
Time = 0.000000



50W 100 cm/s
Time = 0.000276



50W 200 cm/s
Time = 0.000138



Energy deposition rate and defects

50W CW Laser

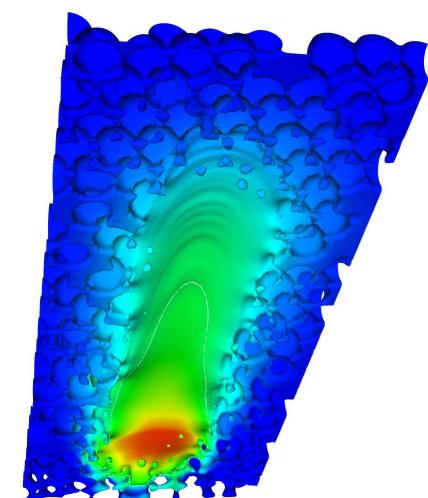
100 cm/s

Time = 0.000350



200 cm/s

Time = 0.000174



- Energy deposition rate $\sim Q/v$
- Higher scan rate
 - Lower energy per unit length
 - Narrower track
 - More void left behind

Part-Scale Models: 304L Cylinder Example

Process

Thermal Model
in Aria

Structure

Microstructure
Model in SPPARKS
(Theron Rodgers)

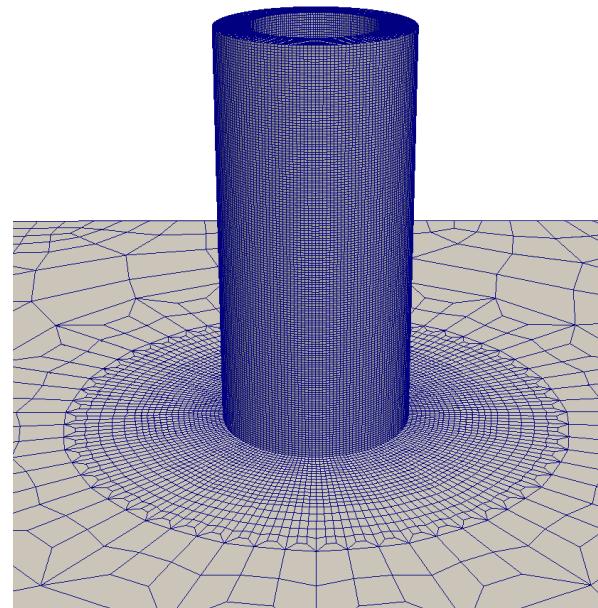
Property

Residual Stress
in Adagio

Performance (Future)

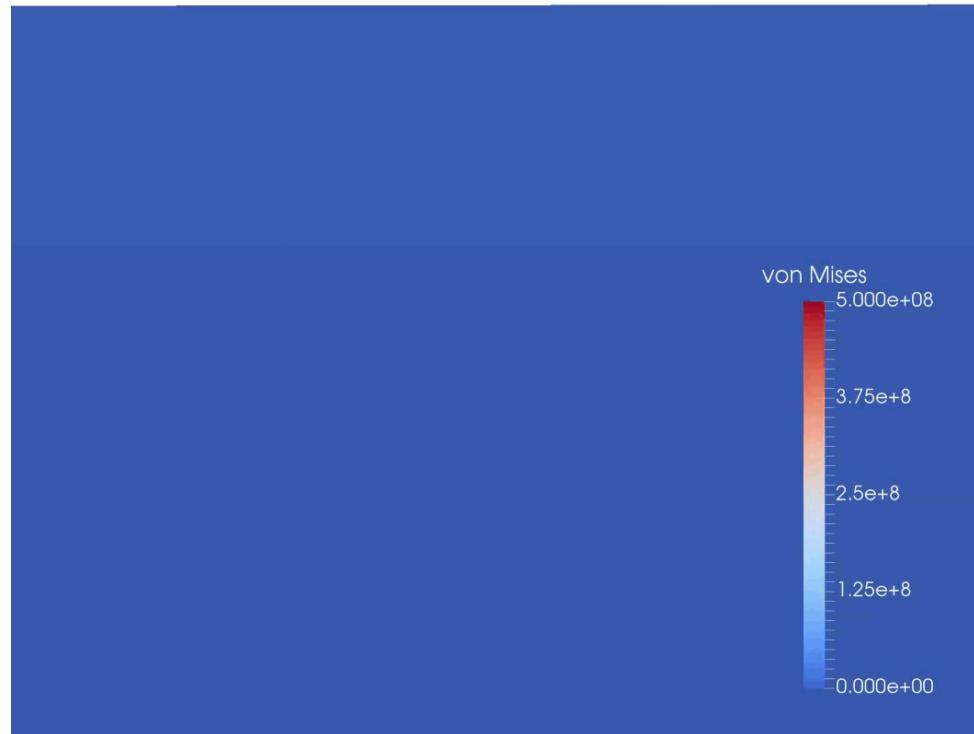
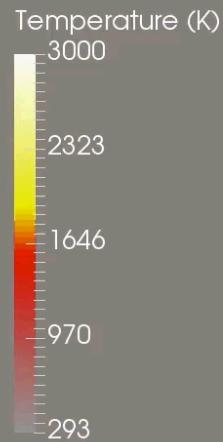
Behavior using as-built
microstructure, residual
stress, and properties

- Cylinder built using LENS process
- Laser diameter = 4 mm
- Laser Speed = 8.46 mm/s
- Layer Thickness = 0.9 mm
- Laser Power = 2000 > 1750 > 1500 > 1250 W

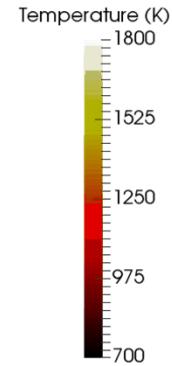
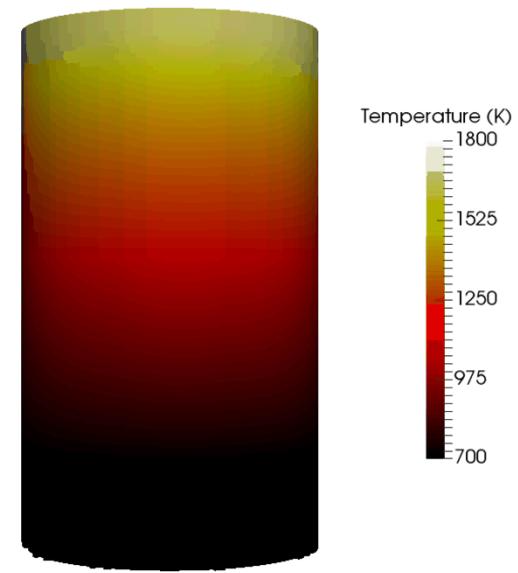
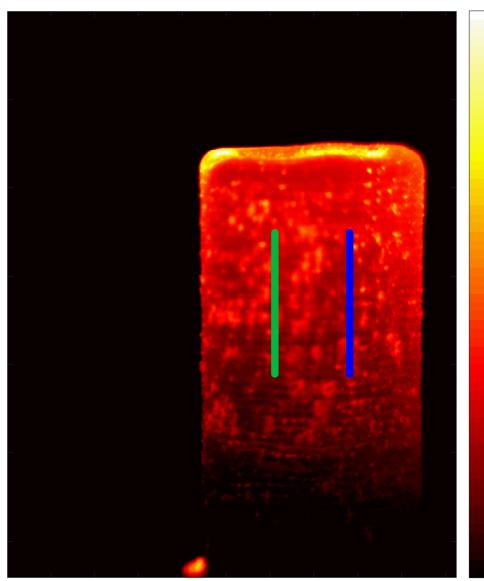
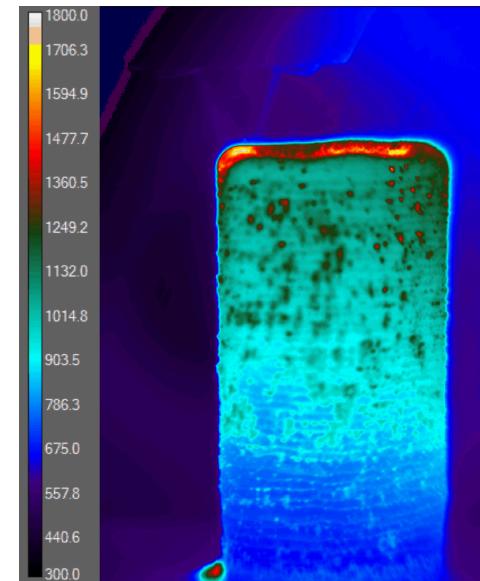


Thermal and Residual Stress Histories

Time: 0.00 s



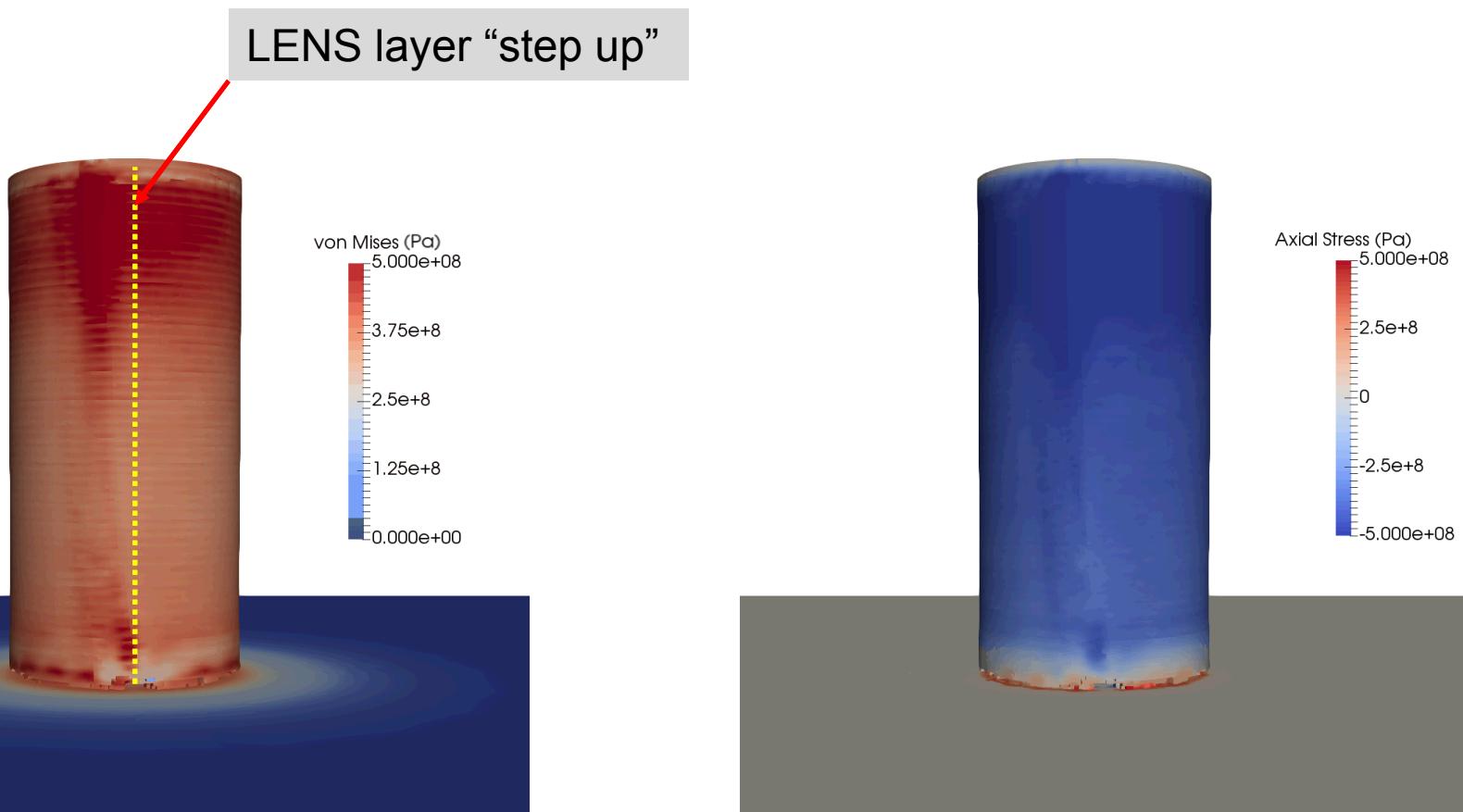
Comparison to IR Imaging



Simulation Results

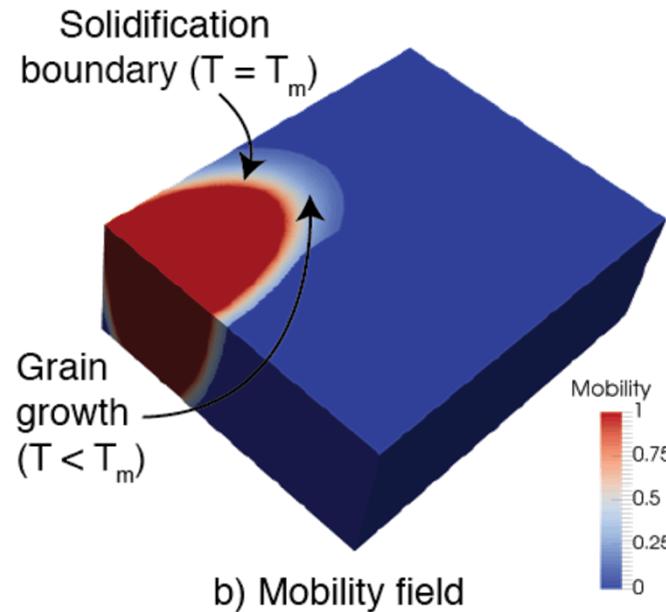
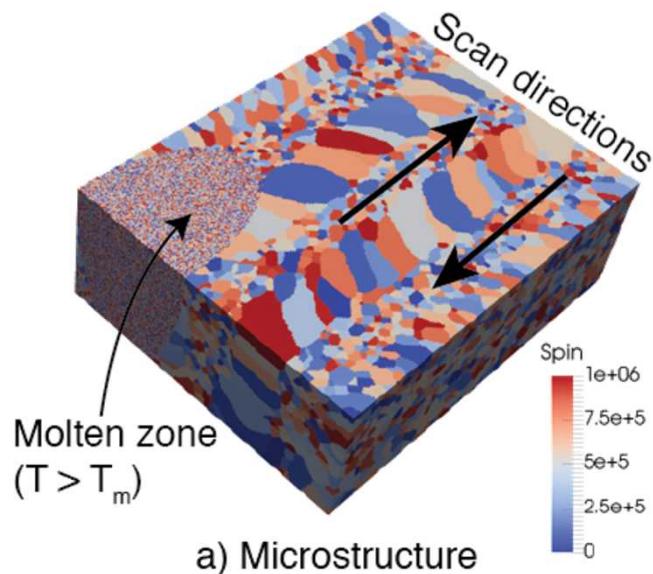
- IR camera mounted on LENS machine
- Assumes constant emissivity
- Compared to simulation

Final Von Mises and Axial Stress



- Images taken after cool-down

Microstructure Prediction in SPPARKS

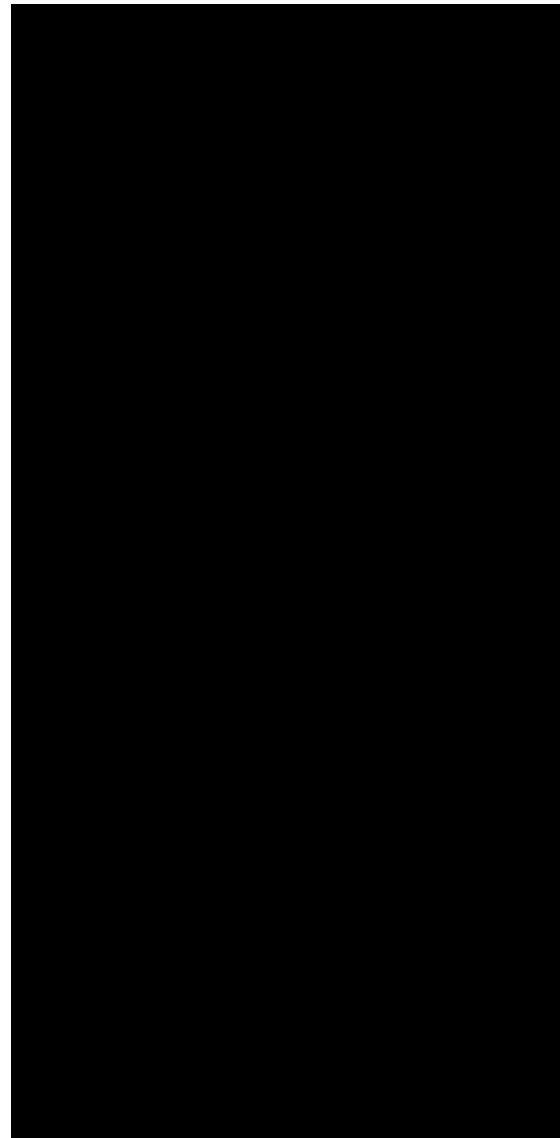


$$M(T) = M_o \exp\left(\frac{-Q}{RT}\right)$$

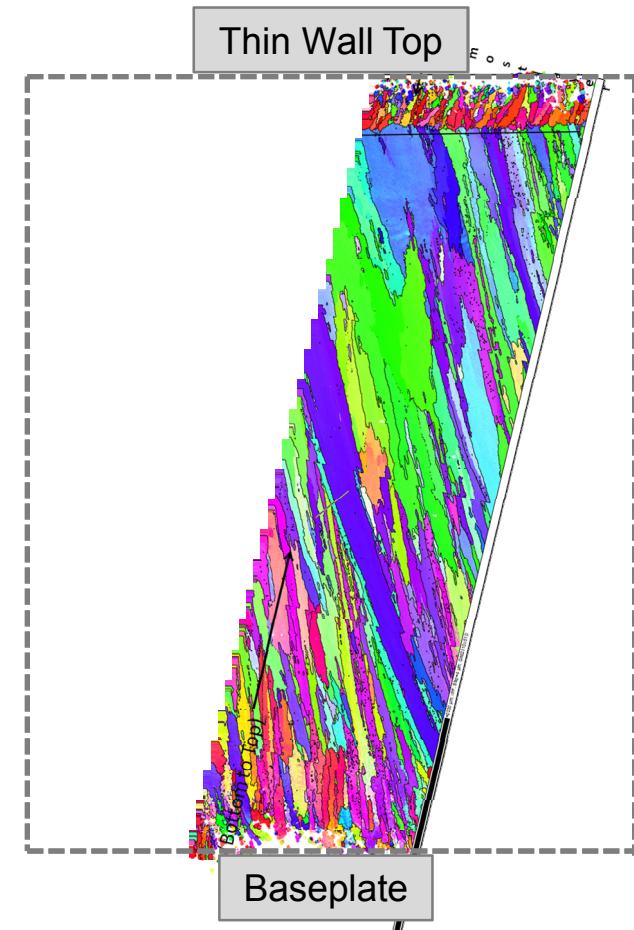
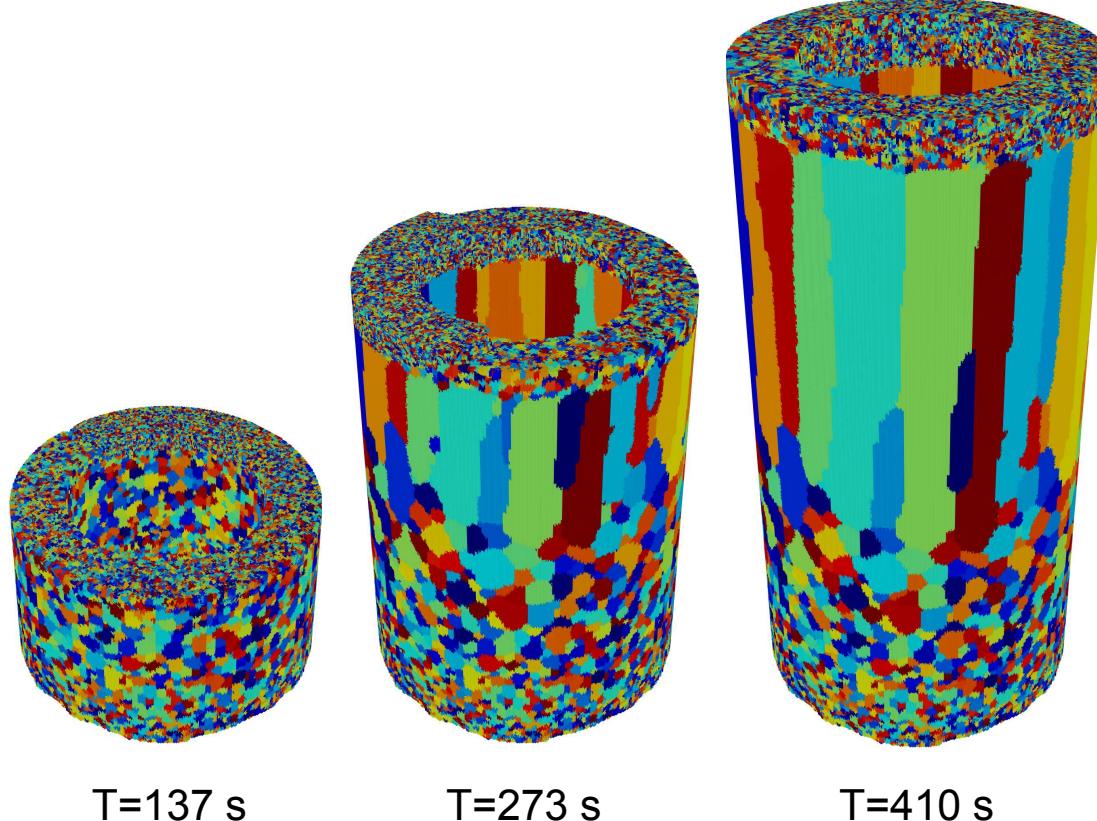
$$P = \begin{cases} M(T) \exp\left(\frac{-\Delta E}{k_B T_s}\right), & \text{if } \Delta E > 0 \\ M(T), & \text{if } \Delta E \leq 0 \end{cases}$$

- Aria temperature history is used as material state in SPPARKS
- Captures bulk heating effects on microstructure

Microstructure Demonstrates Equiaxed to Columnar Grain Transition



Equiaxed to Columnar Transition Observed in Literature

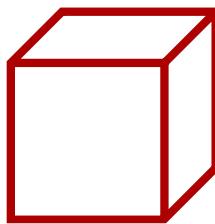


Thin wall IN718 LENS build at 900 W
Parimi *et al.* 2013

Effect of texture on homogenized elastic properties

(J. Bishop)

austenite grain (FCC)



cubic symmetry

$$E = 93.8 \text{ GPa}$$

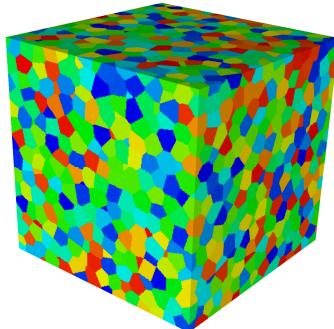
$$\nu = 0.402$$

$$G = 126 \text{ GPa}$$

$$G' \doteq \frac{E}{2(1 + \nu)} = 33.4 \text{ GPa}$$

$$\frac{G}{G'} \approx 3.8 \quad \text{anisotropy ratio}$$

no texture



isotropic

$$E = 198 \text{ GPa}$$

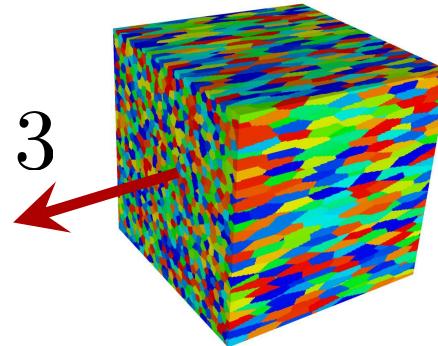
$$\nu = 0.294$$

$$G = 76.5 \text{ GPa}$$

$$G = \frac{E}{2(1 + \nu)}$$

 ideal fiber-texture
along [001]

3



transversely isotropic

$$E_{11} = 143 \text{ GPa}$$

$$E_{22} = 143 \text{ GPa}$$

$$E_{33} = 90.9 \text{ GPa}$$

$$\nu_{12} = 0.114$$

$$\nu_{23} = 0.615$$

$$\nu_{13} = 0.615$$

$$G_{12} = 58 \text{ GPa}$$

$$G_{23} = 126 \text{ GPa}$$

$$G_{13} = 126 \text{ GPa}$$

Summary

- Presented activities and goals for powder bed fusion, thermal-mechanical processing and performance modeling
- Highlighted current capabilities we are leveraging for thermal-mechanical modeling of AM
- SNL is also focusing on impact of UQ, optimization, error estimation for AM