

# Modeling for Metal Additive Manufacturing

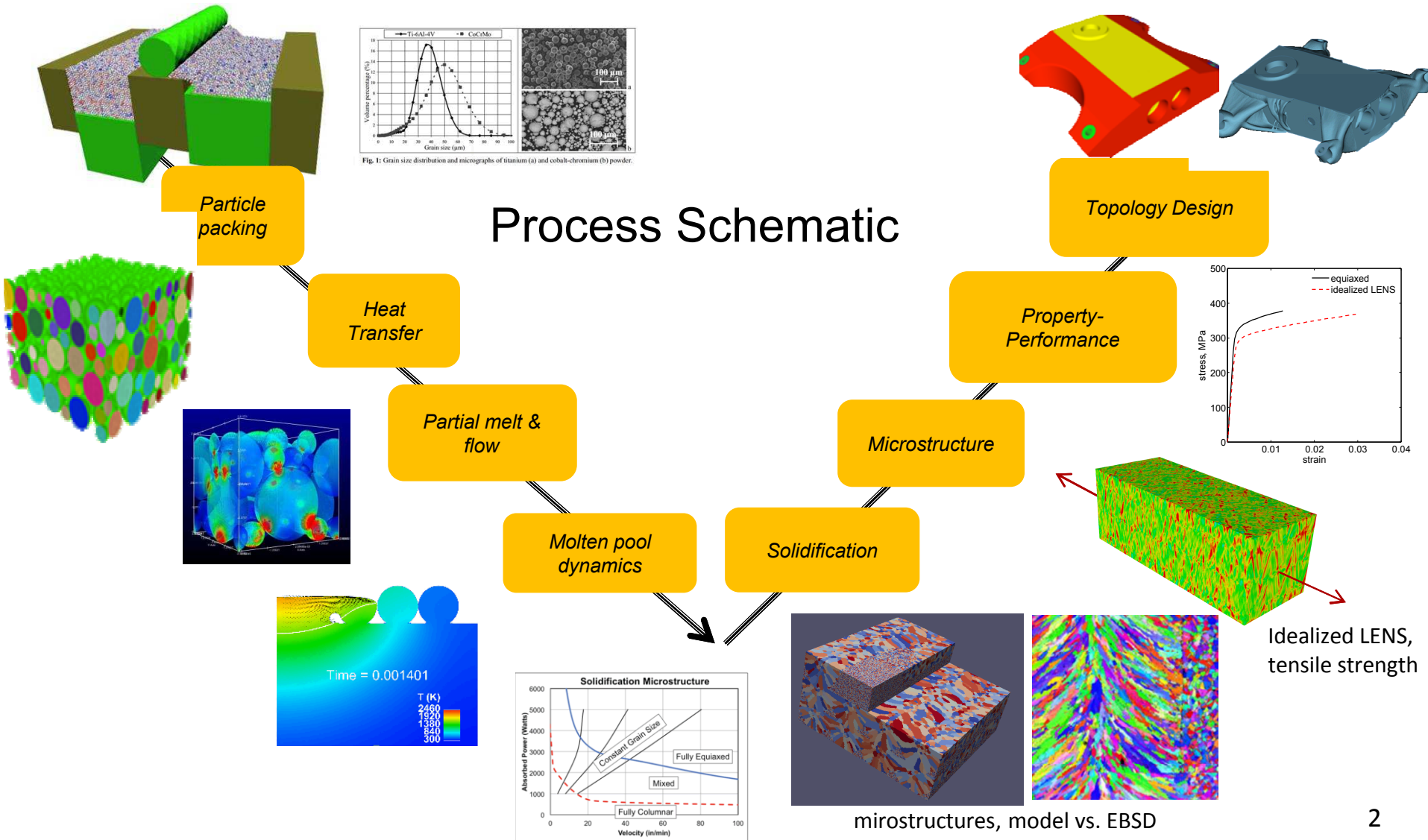
**MJ Martinez and DR Noble**  
Engineering Science Center  
Sandia National Laboratories

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USACM-TMS Additive Manufacturing Conference/Workshop  
Sept. 8, 2017

SAND2017-XXXX

# Modeling Workflow for Additive Manufacturing



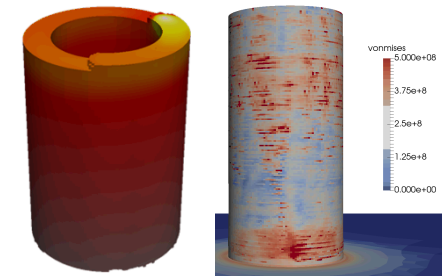
# SNL Modeling Capabilities

## Codes

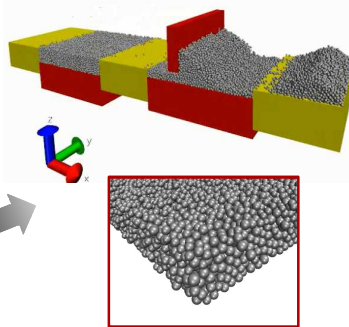
LAMMPS, SPPARKS,  
Sierra/Aria,  
Sierra/Adagio

Part Scale Thermal & Solid Mechanics  
Kyle Johnson, Kurtis Ford & Joe Bishop

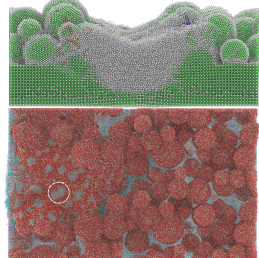
Mesoscale Selective Laser Melting  
Mario Martinez & Brad Trembacki



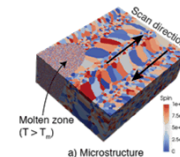
Powder Spreading  
Dan Bolintineanu



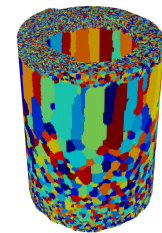
Powder Behavior  
Mark Wilson



Mesoscale Texture/Solid Mechanics/CX  
Judy Brown, Theron Rodgers and Kurtis Ford



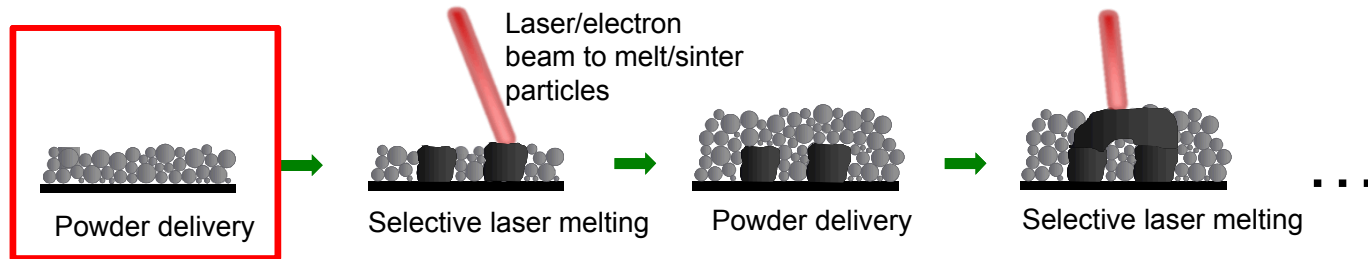
Part Scale Microstructure  
Theron Rodgers



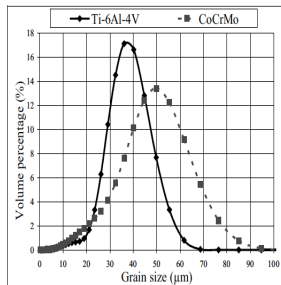
$10^{-6}$   $10^{-3}$  1  
Length Scale (m)

# Powder Dynamics: Background and Motivation

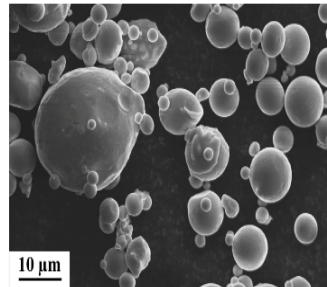
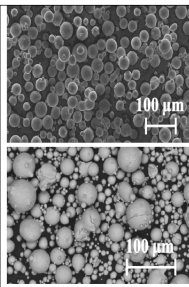
Layer-by-layer powder bed fusion processes (e.g. SLM/SLS):



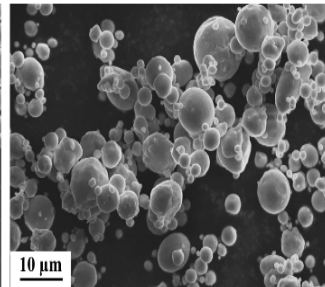
- First step in AM powder bed process
  - Powder bed surface can affect laser interaction; powder bed packing can affect void formation, surface finish, thermal properties
  - Informs downstream process models
- Variability in powder properties due to vendor supply, powder recycling
- Several key process length scales are comparable to individual particles:



From Ref. 1



From Ref. 2



Typical particle diameter: 10-100 μm

Powder layer thickness 30-150 μm

Laser beam spot size 70-200 μm (ref. 1)

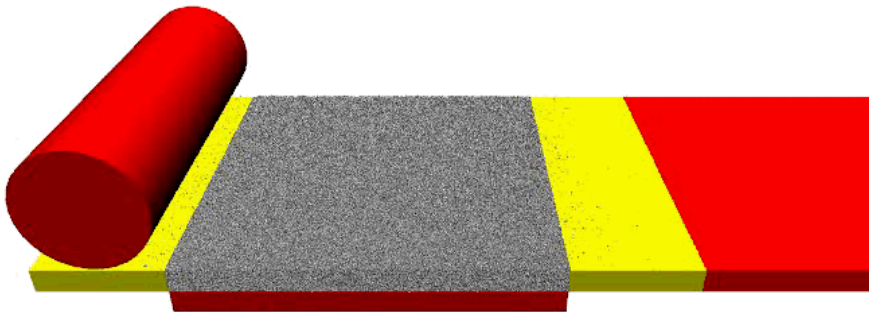
1. Vandenbroucke, B. and Kruth, J.P. *Rapid Prototyping Journal* 13 (2007): 196

2. Yadroitsev, I., et al. *Journal of Laser Applications* 25 (2013): 052003



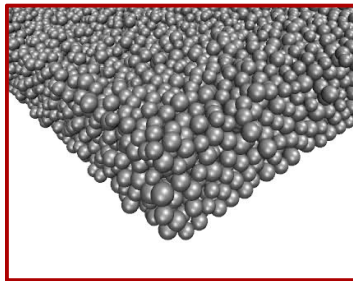
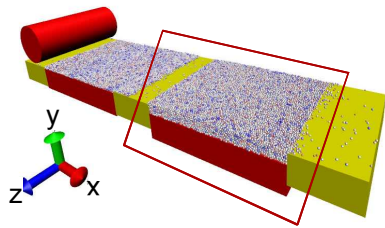
# Modeling powder dynamics using DEM

- **Discrete Element Modeling:** molecular dynamics-like method
  - Each particle modeled explicitly (position, velocity, angular velocity)
  - Forces/torques computed at contact using reduced order models
  - Dynamics integrated in time for large collection of interacting particles



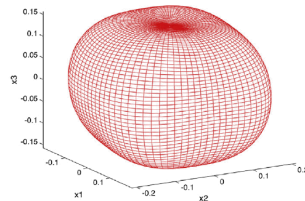
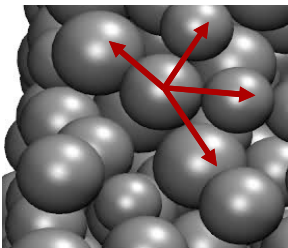
Free, open-source LAMMPS: <https://lammmps.sandia.gov>

# Large resulting data:



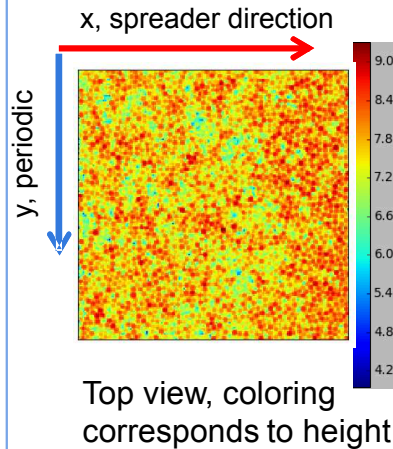
## Local particle contact properties

- Coordination number distribution
- Contact size distribution
- Fabric tensor

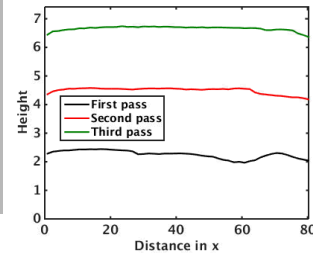


Stershic et al, J Power Sources,  
v. 297, p540

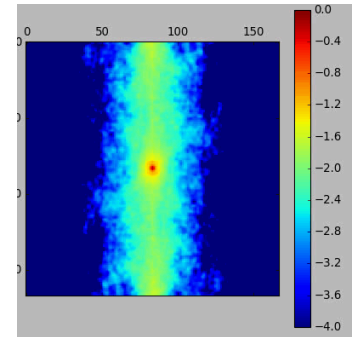
## Powder bed surface properties:



$S_a$ : surface roughness  
Height profile: height averaged over x



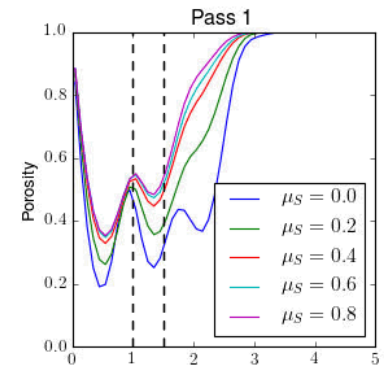
Height autocorrelation function:



## Powder bed bulk properties:

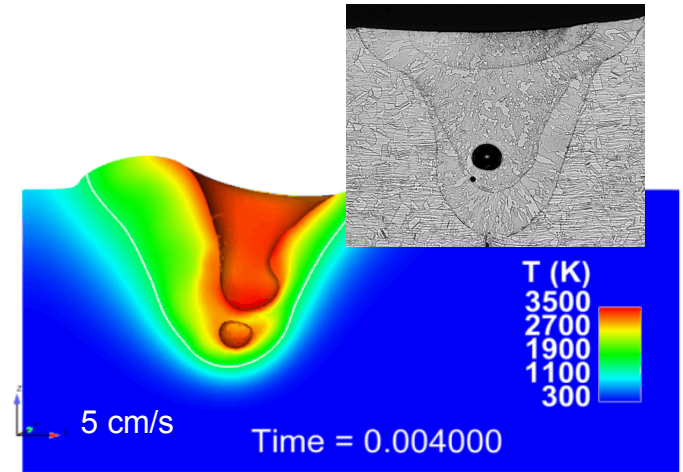
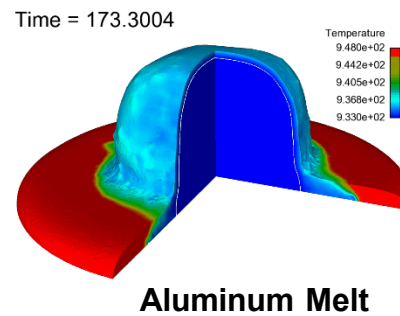
- Porosity(x), porosity(z)
- Pore size distribution
- Two-point correlation function
- 'Coarseness': variability as a function of length scale
- Lineal path function, chord length distribution

...

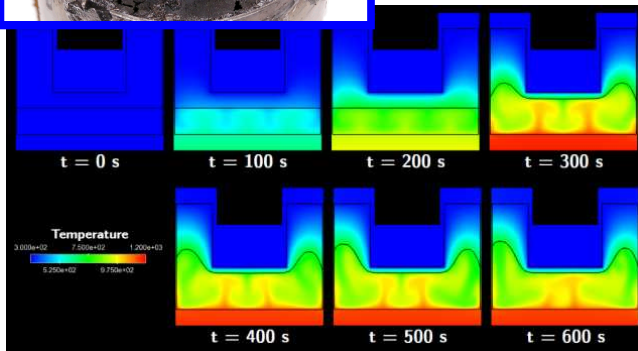


# Relevant Sandia Thermal/Fluid Applications to Burn and Melt using Enriched FEM

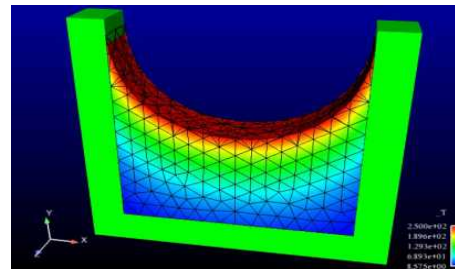
Numerous problems with moving or topologically complex interfaces with discontinuous physics and fields



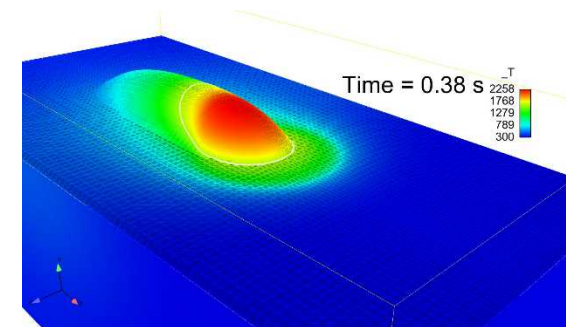
**Laser welding**



**Organic Material Decomposition (OMD)  
with coupled porous and low Ma flow**



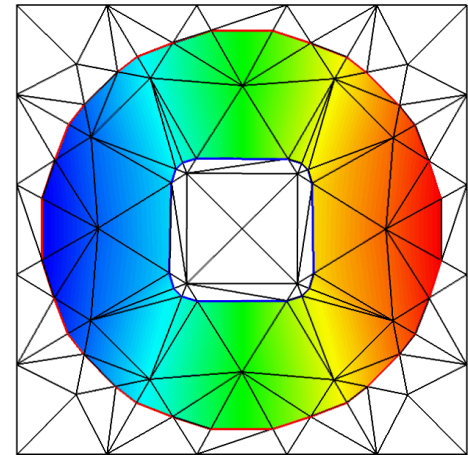
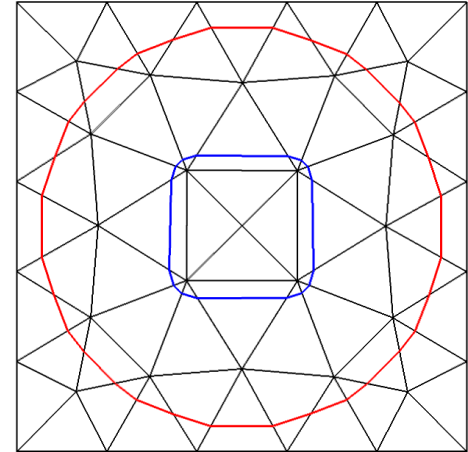
**Material death**



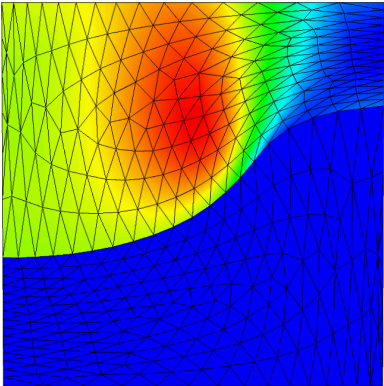
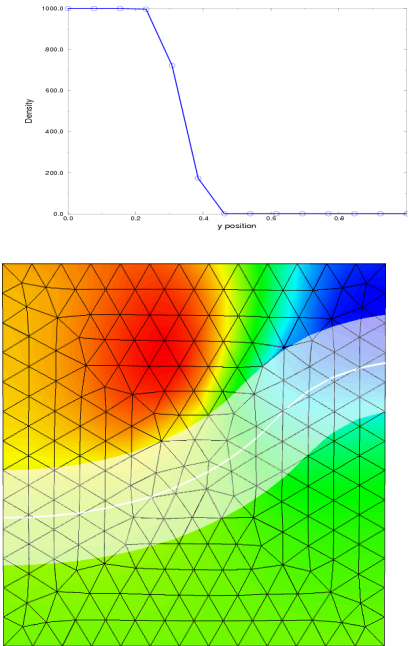
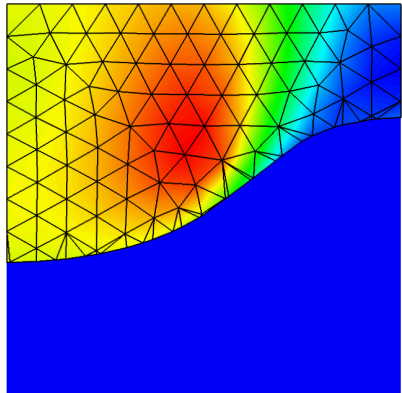
**Additive Manufacturing**

# Conformal Decomposition Finite Element Method (CDFEM)

- Simple Concept (Noble, et al. 2010)
  - Use one or more level set fields to define materials or phases
  - Decompose non-conformal elements into conformal ones
  - Obtain solutions on conformal elements
- Related Work
  - Li et al. (2003) FEM on Cartesian Grid with Added Nodes
  - Ilinca and Hetu (2010) Finite Element Immersed Boundary
  - S. Soghrati and P.H. Geubelle (2012) Interface Enriched Finite Element
- Properties
  - Supports wide variety of interfacial conditions (identical to boundary fitted mesh)
  - Avoids manual generation of boundary fitted mesh
  - Supports general topological evolution (subject to mesh resolution)
- Similar to finite element adaptivity
  - Uses standard finite element assembly including data structures, interpolation, quadrature



# Finite Element Methods for Moving Interfaces in Fluid/Thermal Applications Tested at Sandia

		Enriched Finite Element Methods	
ALE	Diffuse LS	XFEM	CDFEM
<ul style="list-style-type: none"> <li>• Separate, static blocks for gas and liquid phases</li> <li>• Static discretization</li> </ul> 	<ul style="list-style-type: none"> <li>• Single block with smooth transition between gas and liquid phases</li> <li>• Static discretization</li> </ul> 	<ul style="list-style-type: none"> <li>• Single block with sharply enriched elements (weak or strong) spanning gas and liquid phases</li> <li>• Interfacial elements are dynamically enriched to describe phases</li> </ul>	<ul style="list-style-type: none"> <li>• Separate, dynamic blocks for gas and liquid phases</li> <li>• Interfacial elements are dynamically decomposed into elements that conform to phases</li> </ul> 

# Formulation: Interface Dynamics

- Level Set Equation

- Advection equation

$$\frac{\partial \phi}{\partial t} + u \cdot \nabla \phi = 0$$

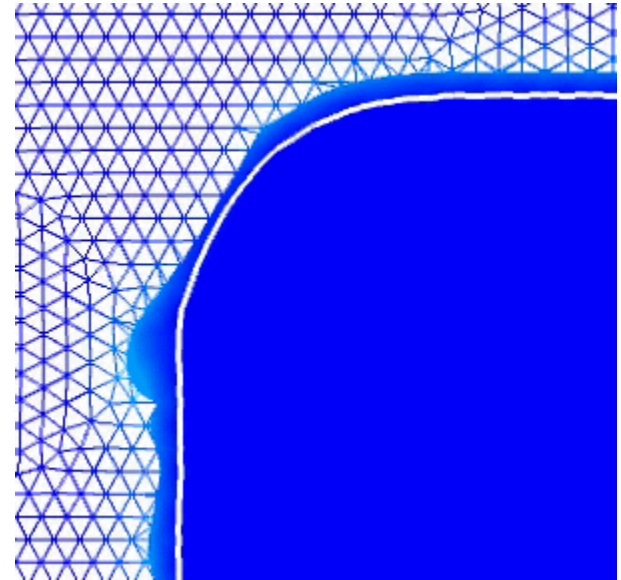
- Galerkin, Backward Euler

$$\int_{\Omega} \frac{\phi - \phi^n}{\Delta t} N_i d\Omega + \int_{\Omega} u \cdot \nabla \phi N_i d\Omega = 0$$

- SUPG stabilization

$$N_i \Rightarrow N_i + \tau_{\phi} u \cdot \nabla N_i, \tau_{\phi} = \left[ \left( \frac{2}{\Delta t} \right)^2 + u_i g_{ij} u_j \right]^{-\frac{1}{2}}$$

- Periodic renormalization
    - Compute nearest distance to interface





# Models: Liquid-Air Interface

## ■ Capillary Force

- Same model used in ALE simulations
  - Jump in stress due to interfacial tension

$$\int_{\Gamma} (\gamma \kappa \mathbf{n} + \nabla_s \gamma) N_i \, d\Gamma = \int_{\Gamma} \gamma \nabla_s N_i \, d\Gamma, \quad \nabla_s \equiv (\mathbf{I} - \mathbf{n}\mathbf{n}) \nabla$$

## ■ Interface Stabilization

- Surface viscosity type stabilization
  - Based on recent paper by Hysing

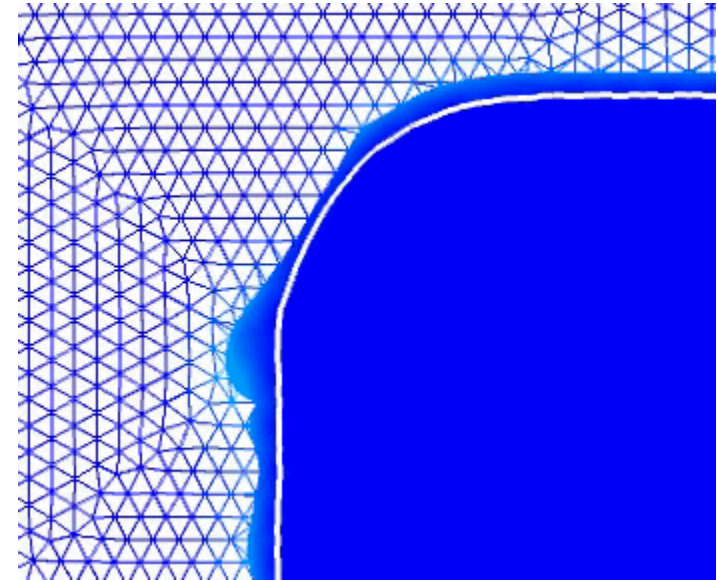
$$\int_{\Gamma} \mu_s \nabla_s u \cdot \nabla N_i \, d\Gamma$$

## ■ Radiation

- Simple radiation boundary condition

$$\int_{\Gamma} \varepsilon \sigma (T^4 - T_e^4) N_i \, d\Gamma$$

- Enclosure radiation
  - Enclosure temperature 2000K
  - Repeat viewfactor calculation every time step

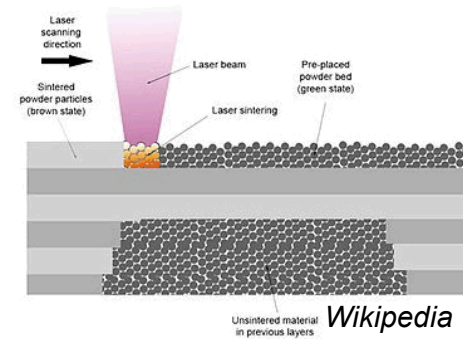
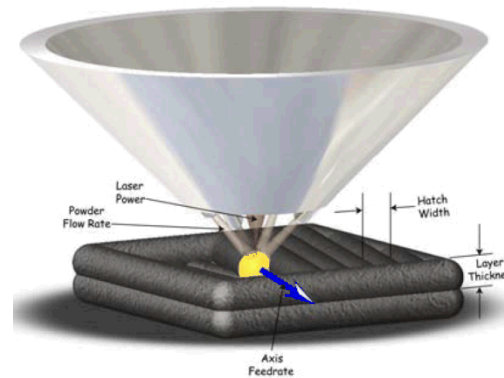


# Powder Bed Physics Model

**Goal:** Link AM mesoscale processes to macroscale performance

**Method:** Conformal level-set technology includes melt and ambient (or assist) gas dynamics

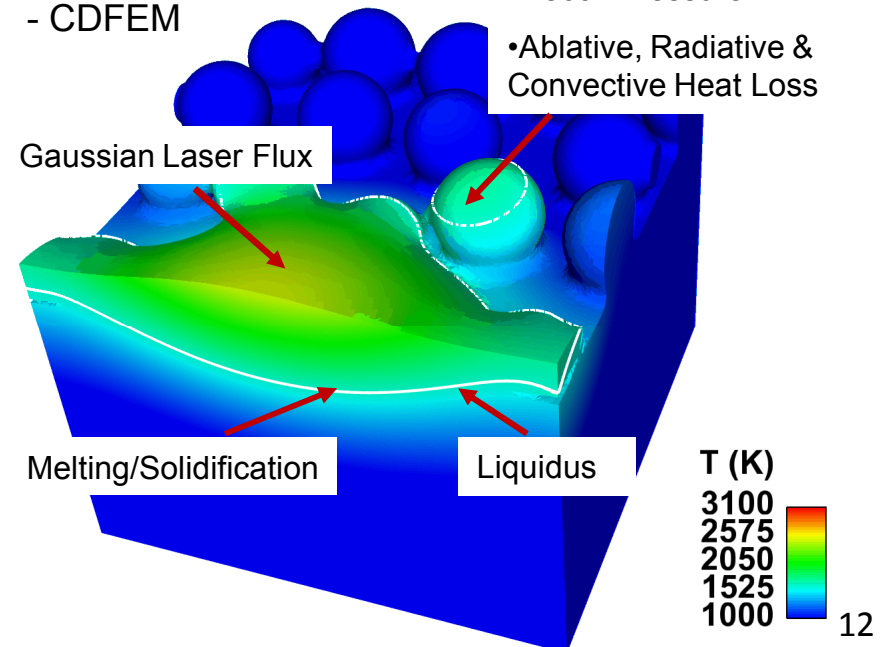
- Laser energy coupling to particle packs
- Melt/solidification, capillary-driven flow, buoyant gas convection, solutal segregation
- Impact of laser setting: power, spot size, scan rate, hatch spacing, ...
- Laser schedule: edge modulation, variable power, variable spot,
- Beam overlap, remelt, porosity



## Free-Surface Motion

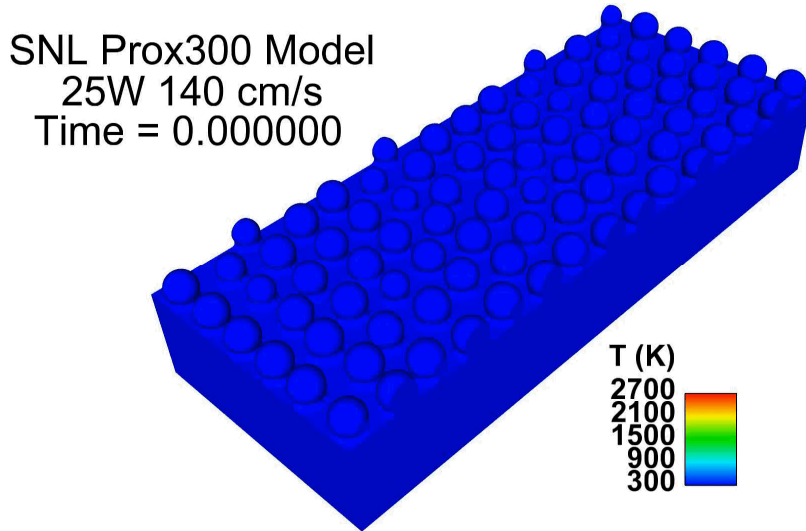
- ALE
- CDFEM

- Curvature & Marangoni Stress
- Recoil Pressure
- Ablative, Radiative & Convective Heat Loss

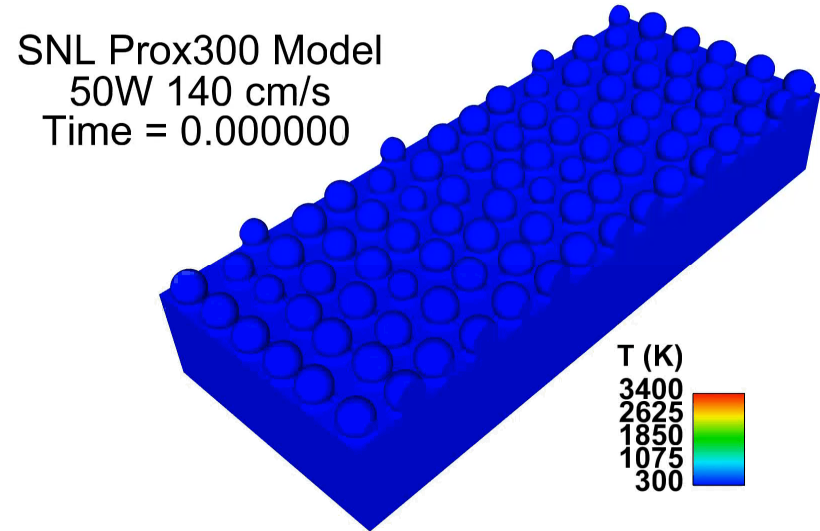


# SNL Prox 300 Model

## Impact of power for fixed hatch spacing

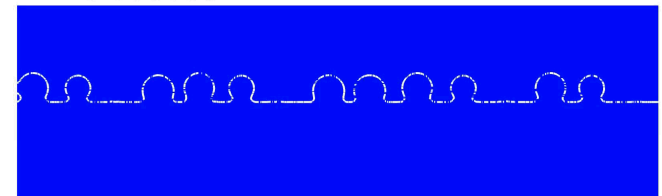


Stainless steel 304L  
25 micron powder



### Gas and melt pool dynamics

Time = 0.000000



T (K)  
3400  
2625  
1850  
1075  
300

#### Notes:

- 500 micron powder bed traversed in **357 microsec.**
- Sloshing-driven **gas dynamics** entrains ambient gas

# SNL Prox 300 Model

## Gaussian Laser 140 cm/s

### General features:

- Surface flux melts particles top to bottom;
- Fully molten particles jump into melt pool by capillarity, inciting sloshing

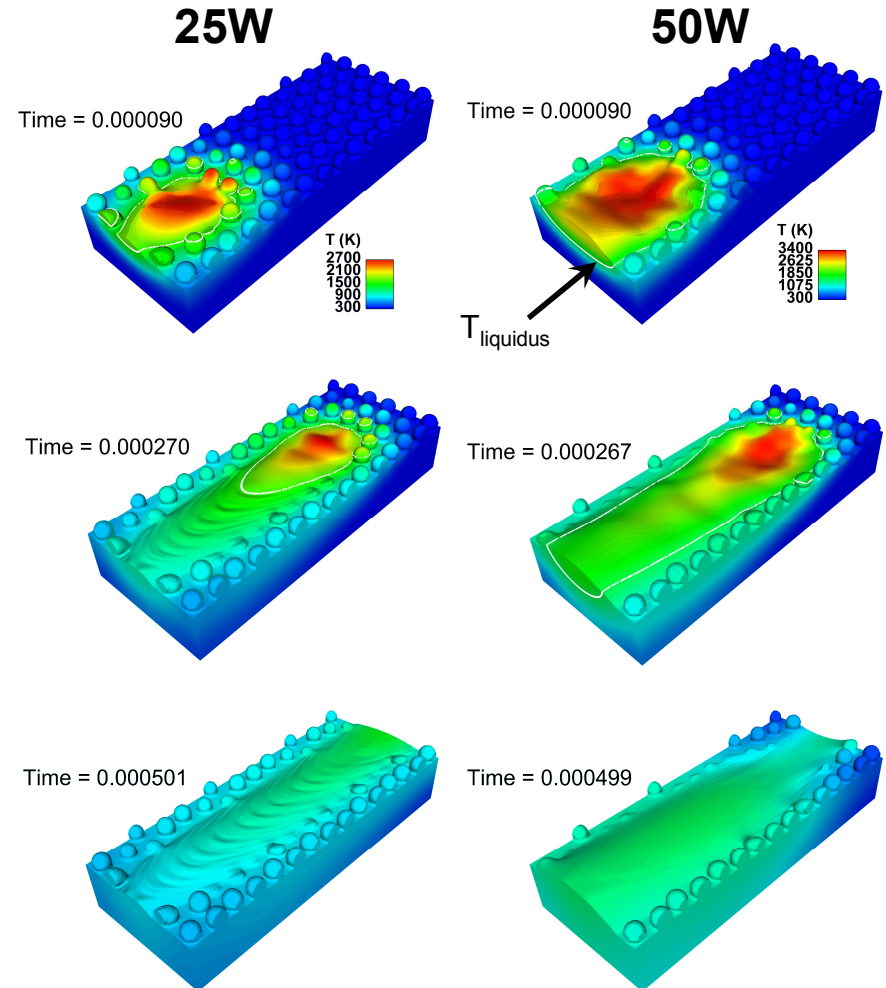
### 25W

- Shallow melt pool; smaller melt path
- Solidification front freezes in wave peaks and troughs – *ribbed finish*

### 50W

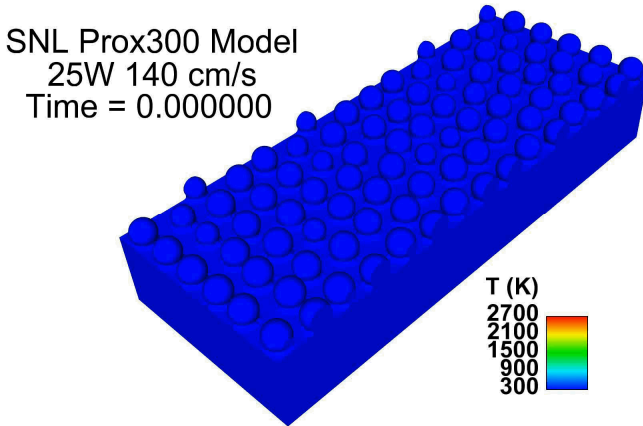
- Long, deep melt pool; wider melt path
- *Smooth finish* (?)
- Laser power modulated at exit

***Power, hatch spacing, and vertical heat penetration can be guided/optimized with melt dynamics models.***

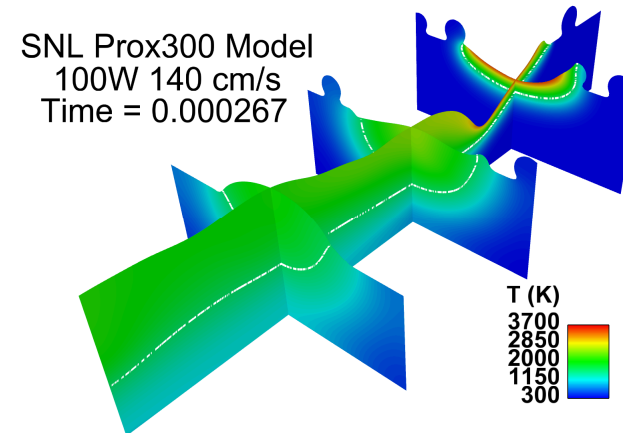
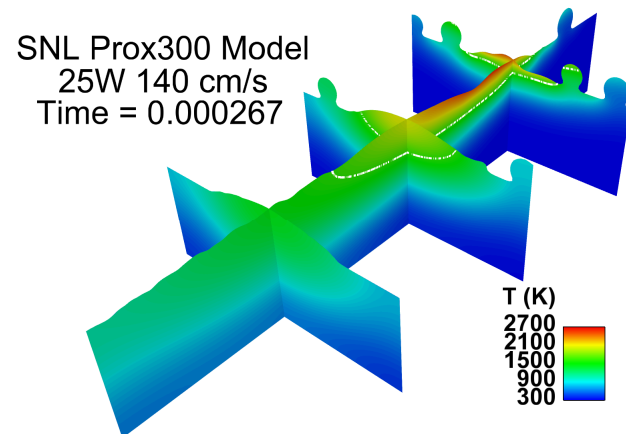
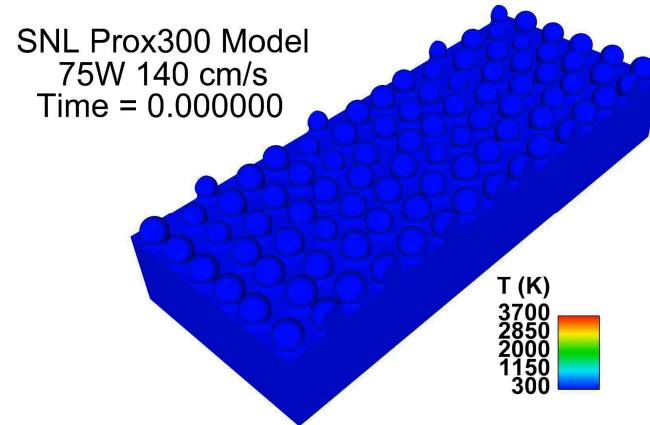


# Power and recoil pressure

25W



100 W



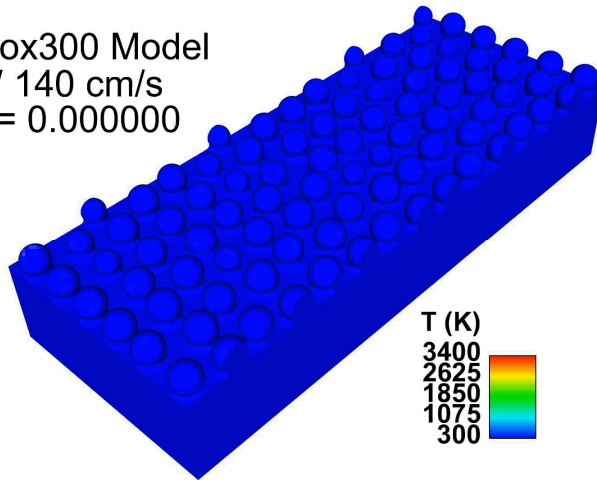
- Small melt pool
- Freezes in surface ripples

- Keyhole (small) at high power
- Recoil pressure → balling effect

# Powder porosity

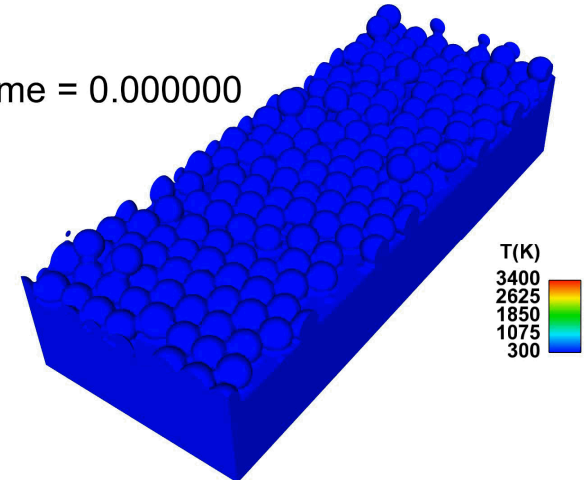
50W Reff=0.6

SNL Prox300 Model  
50W 140 cm/s  
Time = 0.000000



Sparse powder bed  
27% solid fraction

Time = 0.000000



Simulated powder bed  
57% solid powder fraction

Lower porosity powder:

- requires more laser power
- Higher void probability

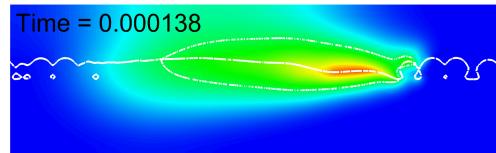
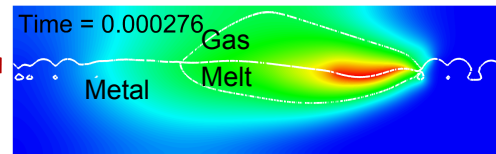
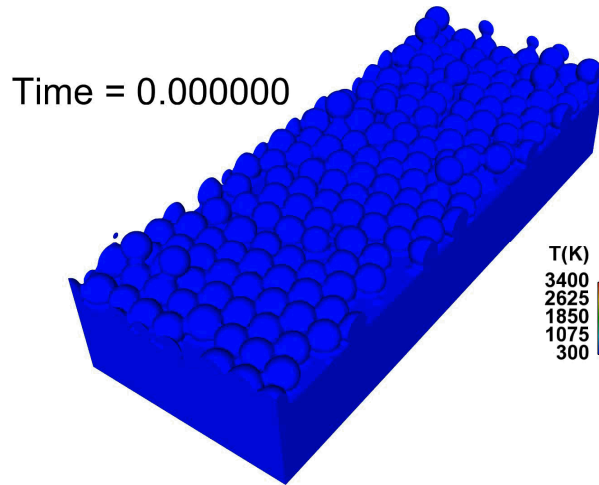


# Laser Scan Rate

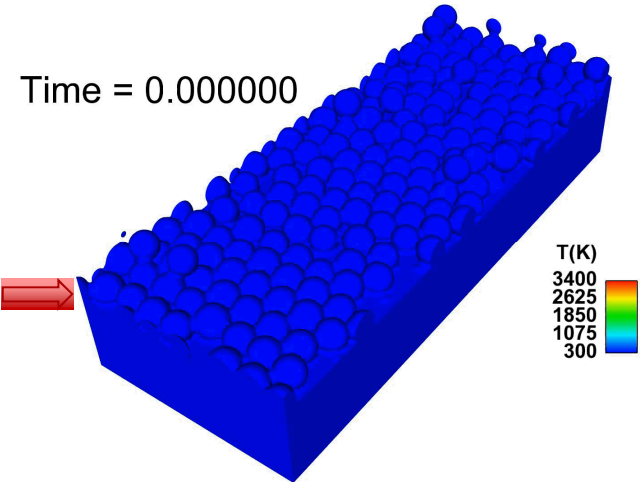
Energy deposition rate  $\sim Q/v$

50W Reff=0.6

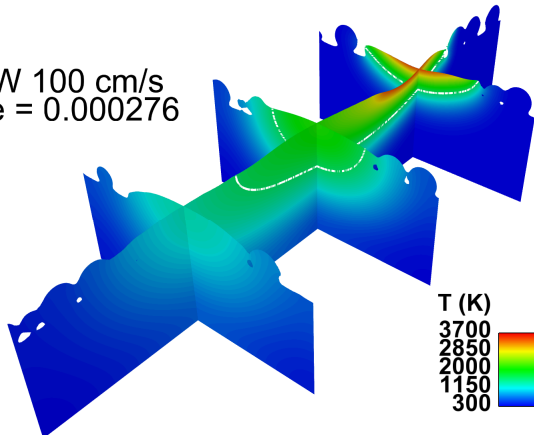
100 cm/s



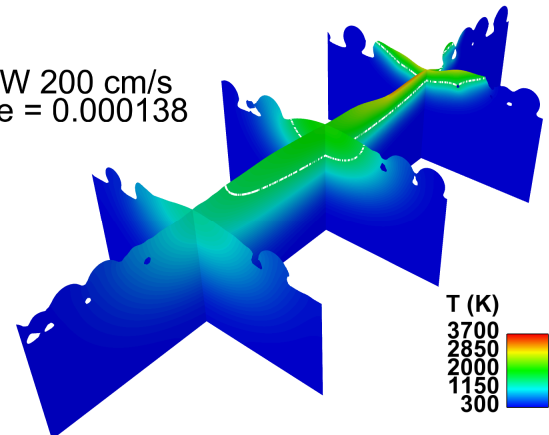
200 cm/s



50W 100 cm/s  
Time = 0.000276



50W 200 cm/s  
Time = 0.000138



# Energy deposition rate and defects

## 50W CW Laser

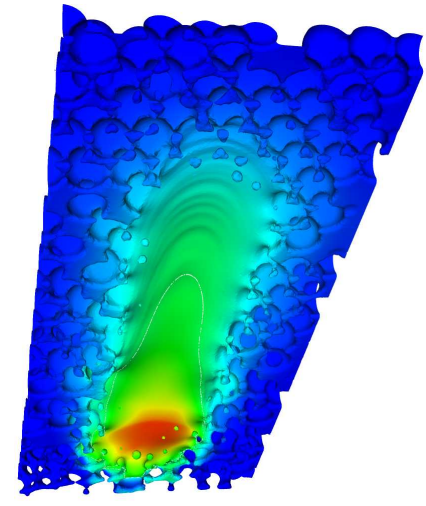
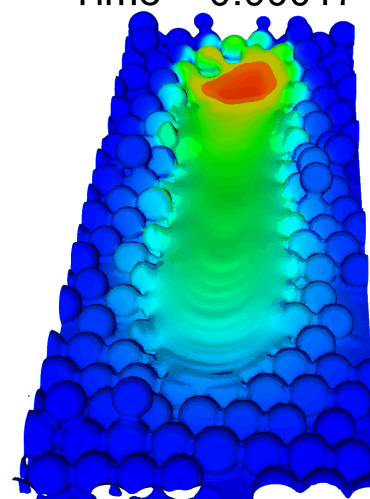
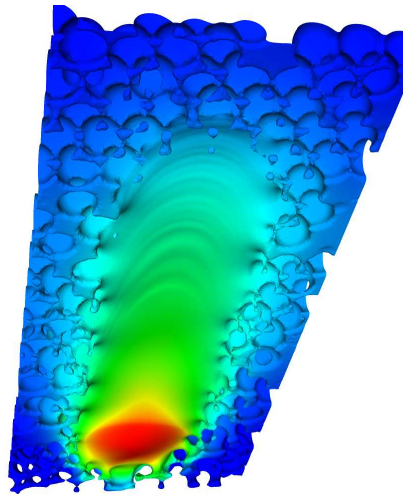
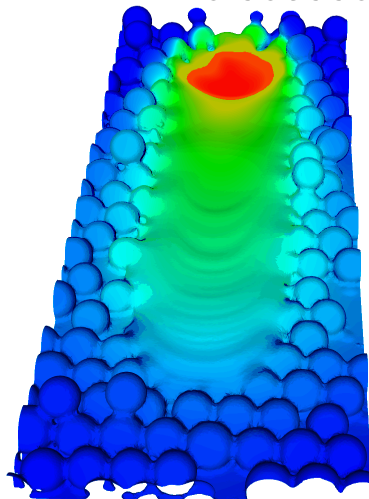
100 cm/s

200 cm/s

Time = 0.000350

Time = 0.000174

T(K)  
3400  
2625  
1850  
1075  
300



- Energy deposition rate  $\sim Q/v$
- Higher scan rate
  - Lower energy per unit length
  - Narrower track
  - More void left behind

# Part-Scale Models: 304L Cylinder Example

## Process

Thermal Model  
in Aria

## Structure

Microstructure  
Model in SPPARKS  
(Theron Rodgers)

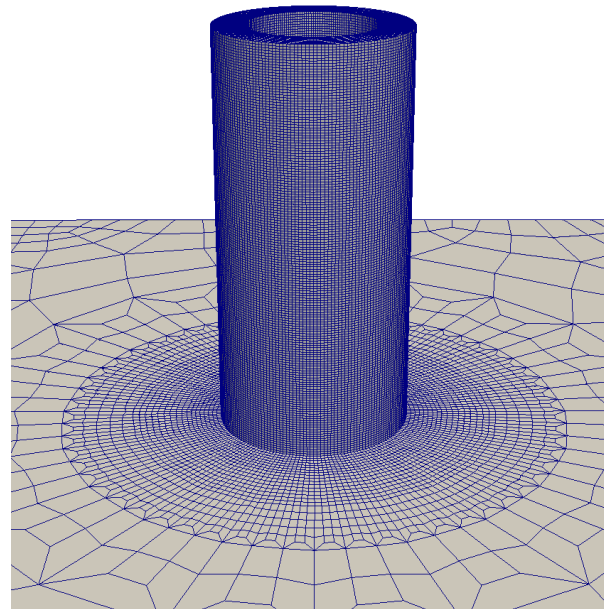
## Property

Residual Stress  
in Adagio

## Performance (Future)

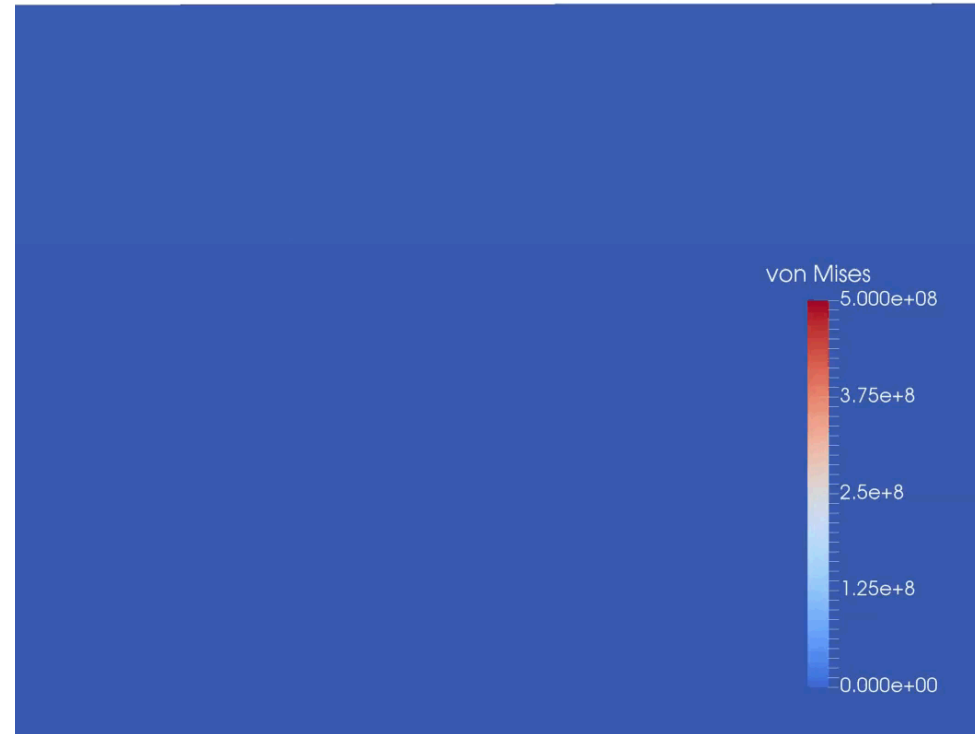
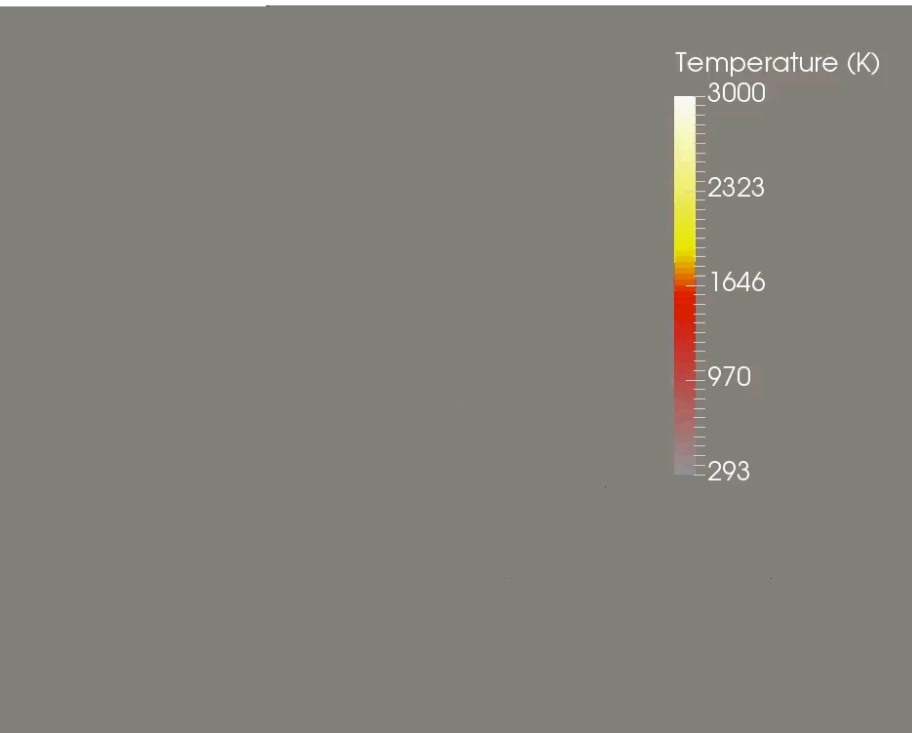
Behavior using as-  
built microstructure,  
residual stress, and  
properties

- Cylinder built using LENS process
- Laser diameter = 4 mm
- Laser Speed = 8.46 mm/s
- Layer Thickness = 0.9 mm
- Laser Power = 2000 > 1750 > 1500 > 1250 W

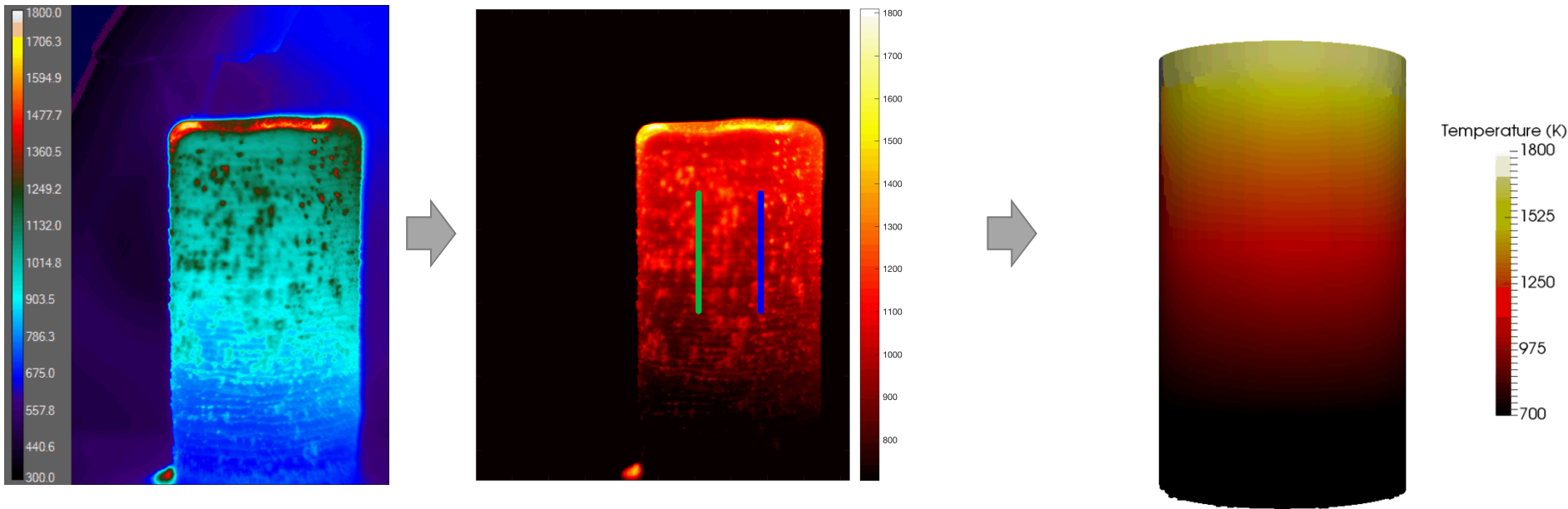


# Thermal and Residual Stress Histories

Time: 0.00 s



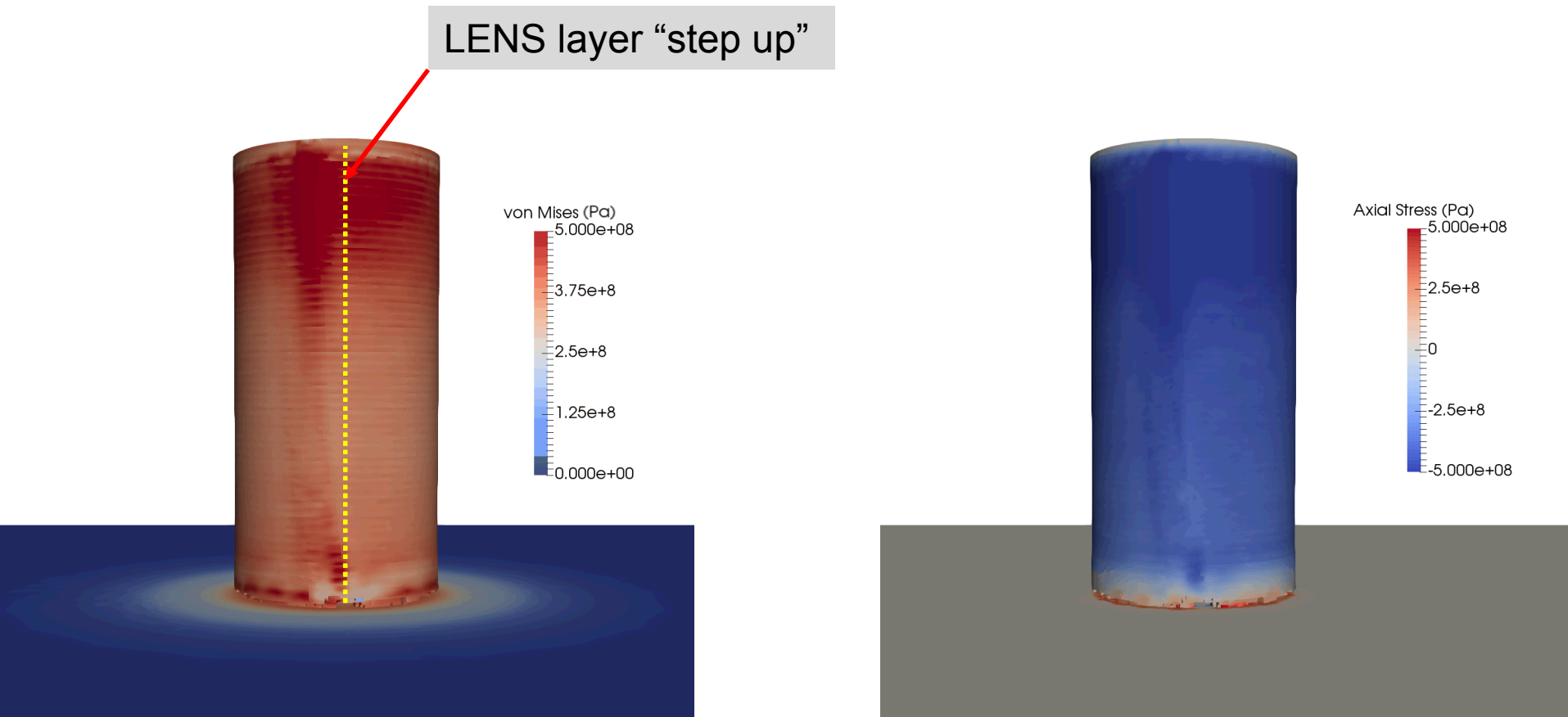
# Comparison to IR Imaging



Simulation  
Results

- IR camera mounted on LENS machine
- Assumes constant emissivity
- Compared to simulation

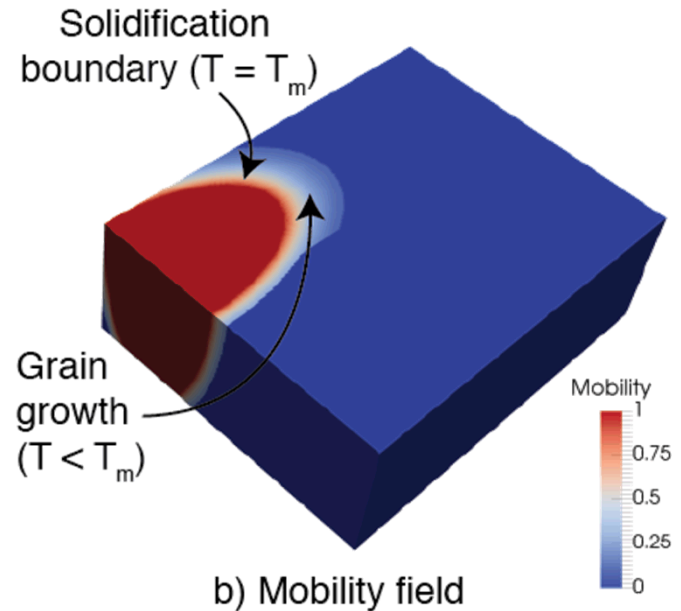
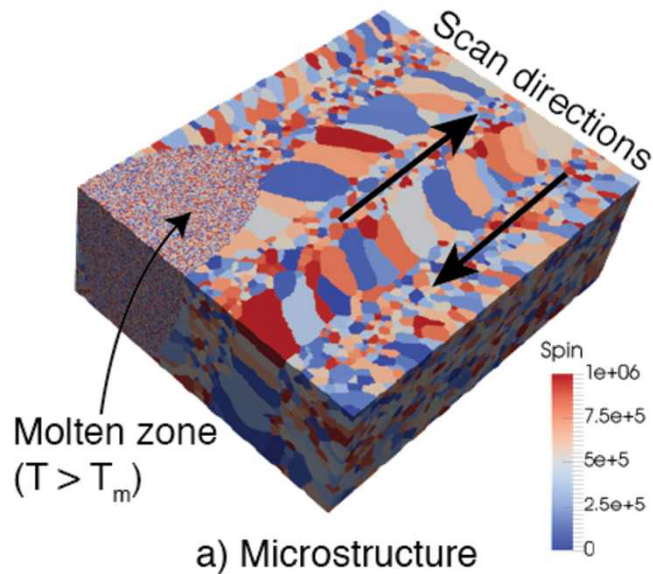
# Final Von Mises and Axial Stress



- Images taken after cool-down



# Microstructure Prediction in SPPARKS

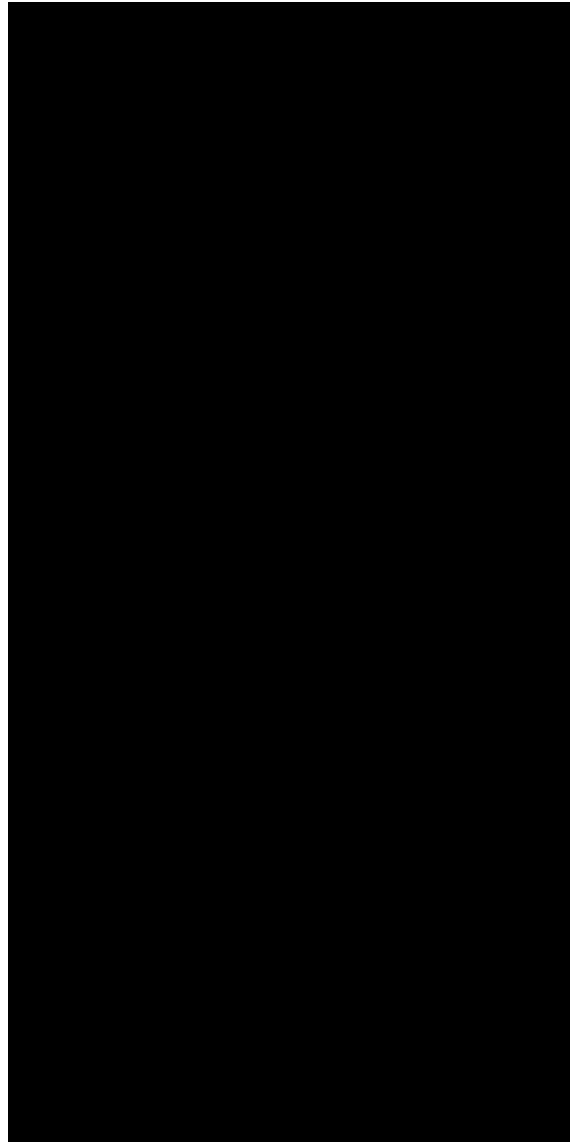


$$M(T) = M_o \exp\left(\frac{-Q}{RT}\right)$$

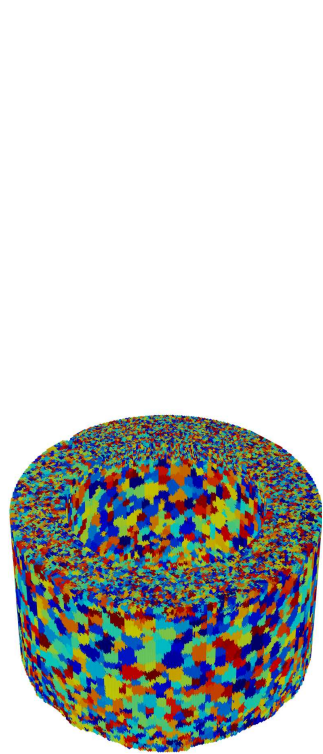
$$P = \begin{cases} M(T) \exp\left(\frac{-\Delta E}{k_B T_s}\right), & \text{if } \Delta E > 0 \\ M(T), & \text{if } \Delta E \leq 0 \end{cases}$$

- Aria temperature history is used as material state in SPPARKS
- Captures bulk heating effects on microstructure

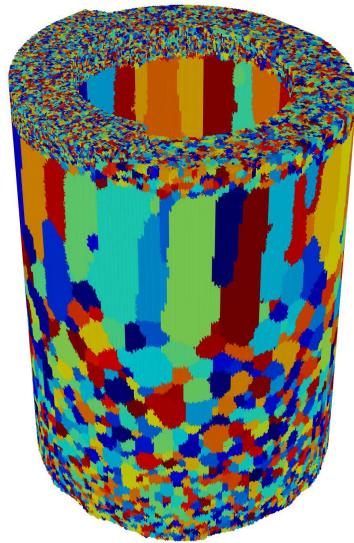
# Microstructure Demonstrates Equiaxed to Columnar Grain Transition



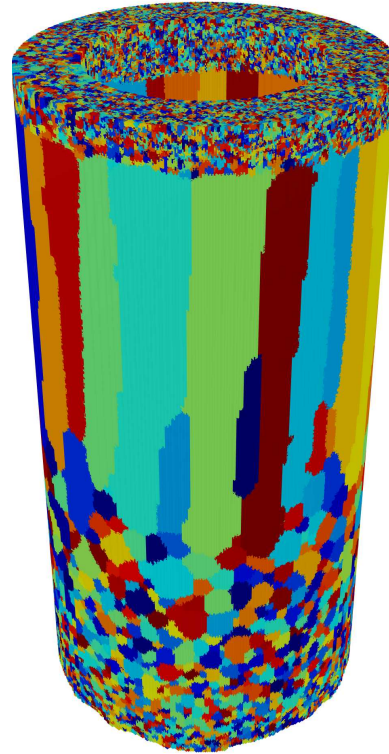
# Equiaxed to Columnar Transition Observed in Literature



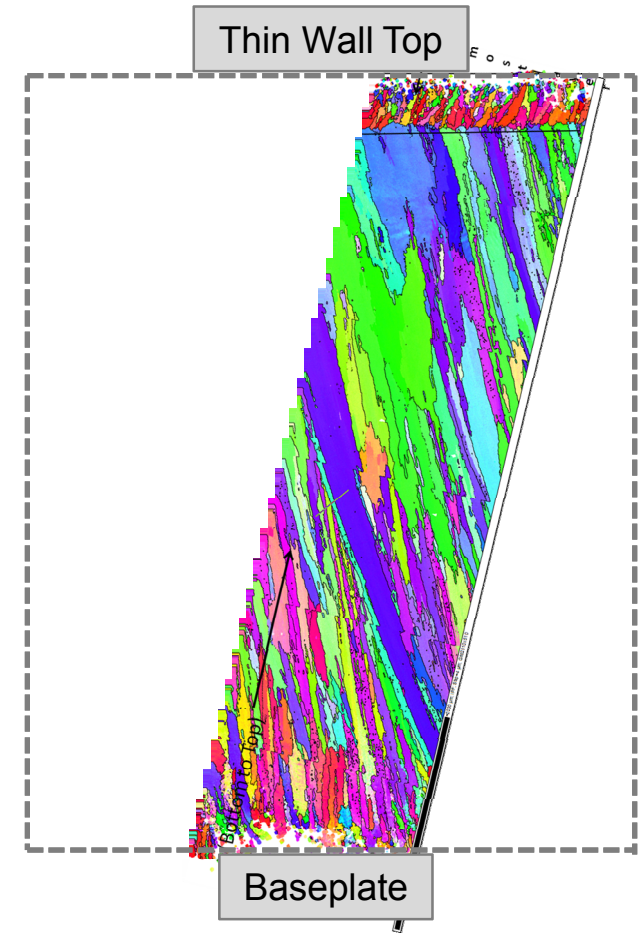
T=137 s



T=273 s



T=410 s

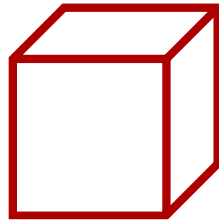


Thin wall IN718 LENS build at 900 W  
Parimi *et al.* 2013

# Effect of texture on homogenized elastic properties

(J. Bishop)

austenite grain (FCC)



cubic symmetry

$$E = 93.8 \text{ GPa}$$

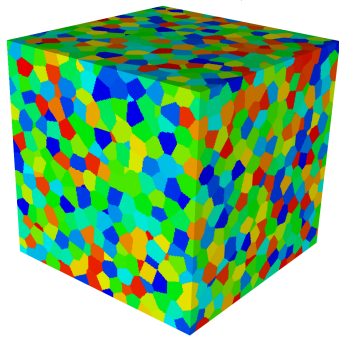
$$\nu = 0.402$$

$$G = 126 \text{ GPa}$$

$$G' \doteq \frac{E}{2(1 + \nu)} = 33.4 \text{ GPa}$$

$$\frac{G}{G'} \approx 3.8 \text{ anisotropy ratio}$$

no texture



isotropic

$$E = 198 \text{ GPa}$$

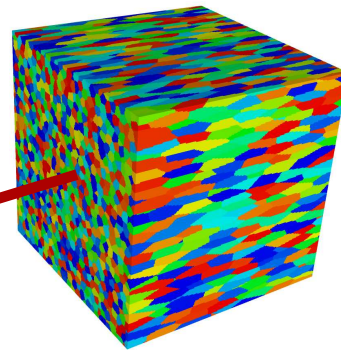
$$\nu = 0.294$$

$$G = 76.5 \text{ GPa}$$

$$G = \frac{E}{2(1 + \nu)}$$

ideal fiber-texture  
along [001]

3



transversely isotropic

$$E_{11} = 143 \text{ GPa}$$

$$E_{22} = 143 \text{ GPa}$$

$$E_{33} = 90.9 \text{ GPa}$$

$$\nu_{12} = 0.114$$

$$\nu_{23} = 0.615$$

$$\nu_{13} = 0.615$$

$$G_{12} = 58 \text{ GPa}$$

$$G_{23} = 126 \text{ GPa}$$

$$G_{13} = 126 \text{ GPa}$$

# Summary

- Presented activities and goals for powder bed fusion, thermal-mechanical processing and performance modeling
- Highlighted current capabilities we are leveraging for thermal-mechanical modeling of AM
- SNL is also focusing on impact of UQ, optimization, error estimation for AM