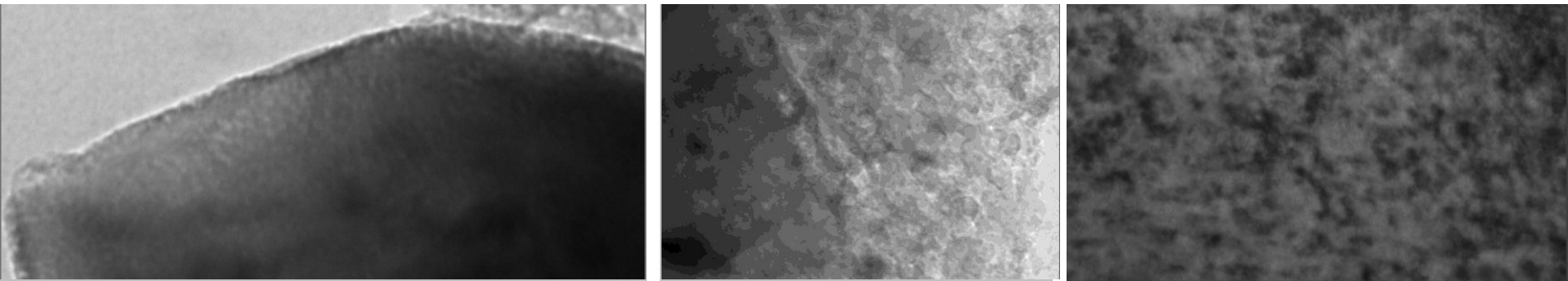


*Exceptional service in the national interest*

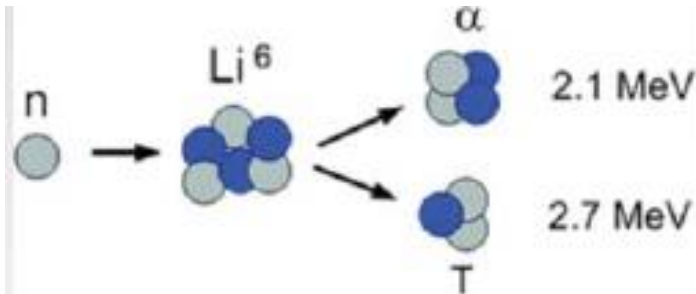


## **Simulations of Damage and Gas Accumulation in TPBAR Materials with In-situ Triple Ion Beam Irradiation TEM**

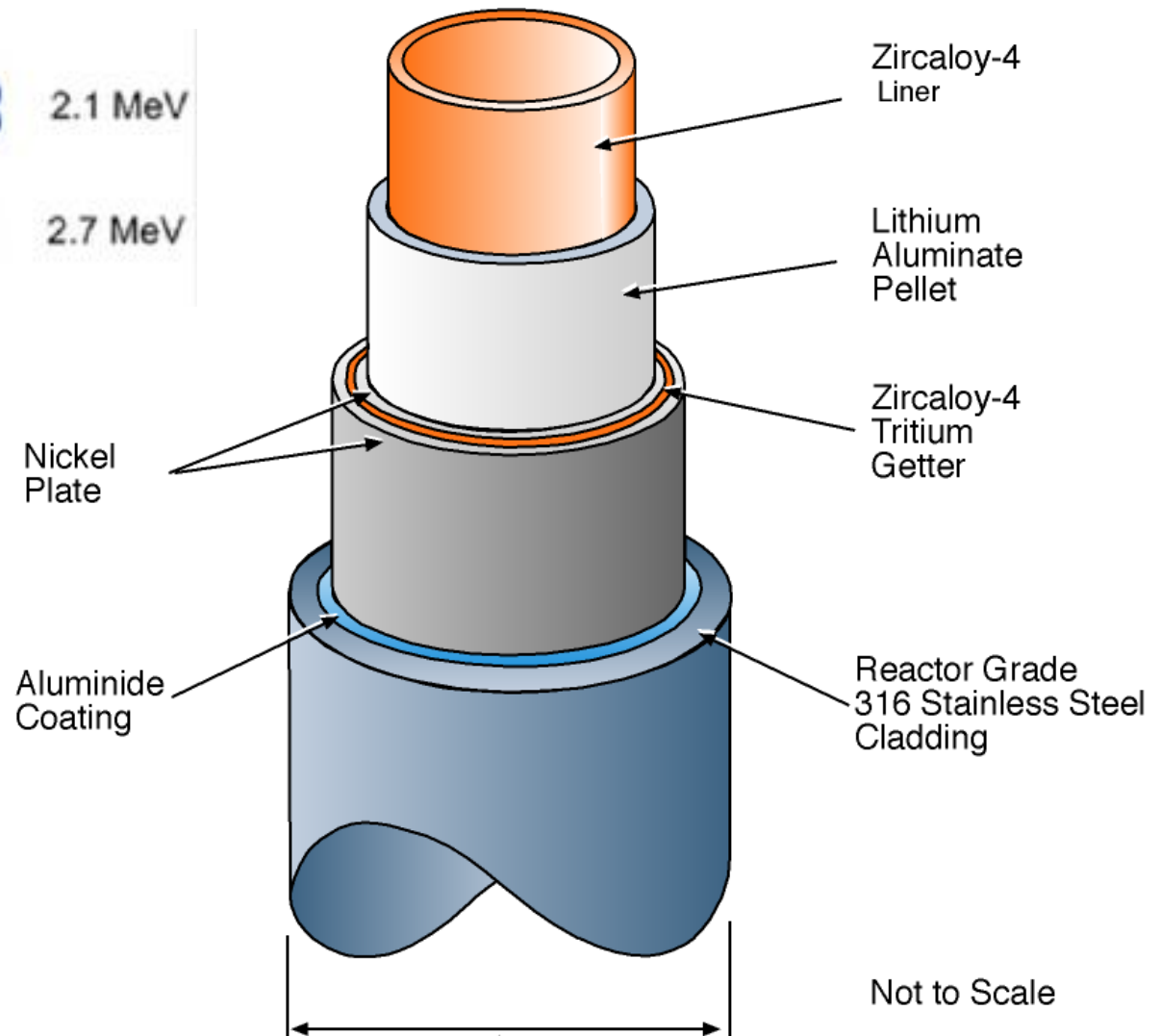
Caitlin A. Taylor, Brittany Muntifering, David Senior, Clark Snow, and Khalid Hattar

**September 6<sup>th</sup>, 2017**

# Tritium Producing Burnable Absorber Rod

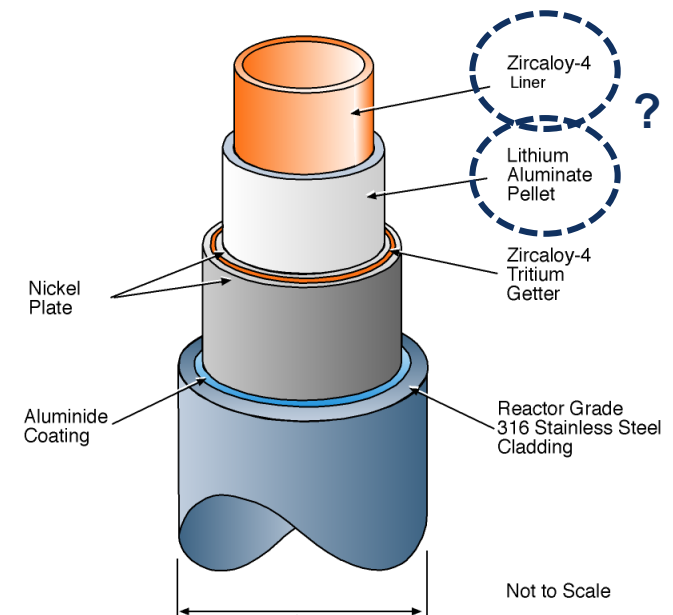
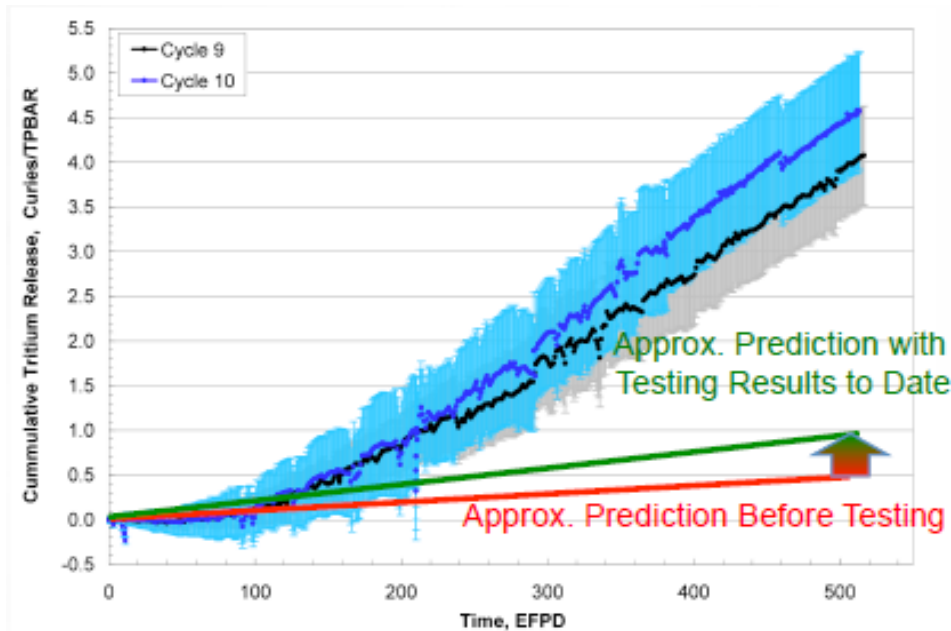


- Displacement Damage
- Helium Implantation
- Tritium Implantation
- Elevated Temperatures



# Understanding Tritium Permeation in TPBAR

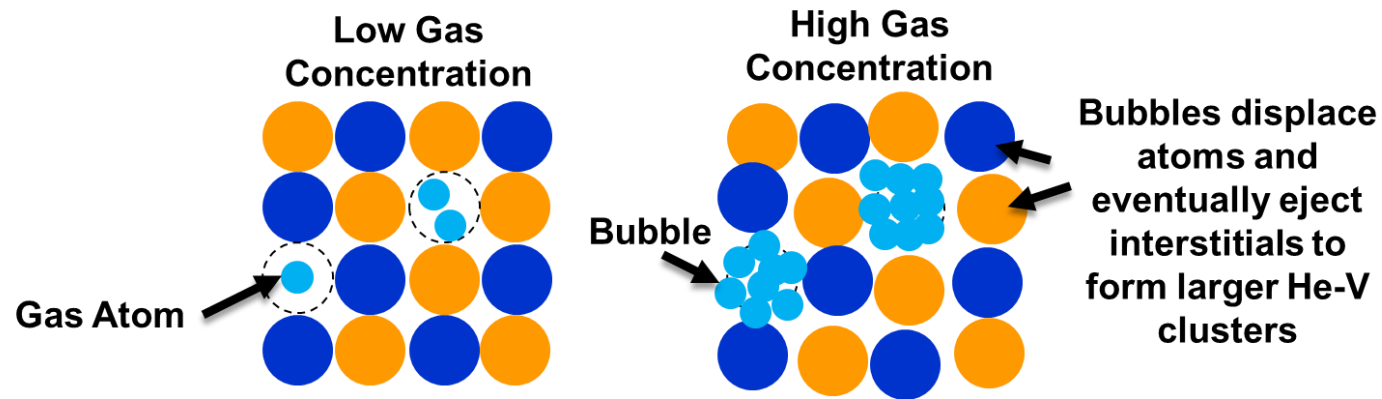
- TPBAR  $^3\text{H}$  permeation is higher than predictive performance models
  - In 2004, during Cycle 6, the predicted levels were  $\sim 0.5$  Ci/TPBAR/cycle and actual levels were  $\sim 4$  Ci/TPBAR/cycle (0.04% of total  $^3\text{H}$  produced)
- Mechanisms responsible for differences between predictions and observations are not well understood
- Currently building an understanding of fundamental  $^3\text{H}$ -He-defect interactions



Burkes, Senior, Longoni and Johns, TFG Meeting  
2016, Rochester, NY

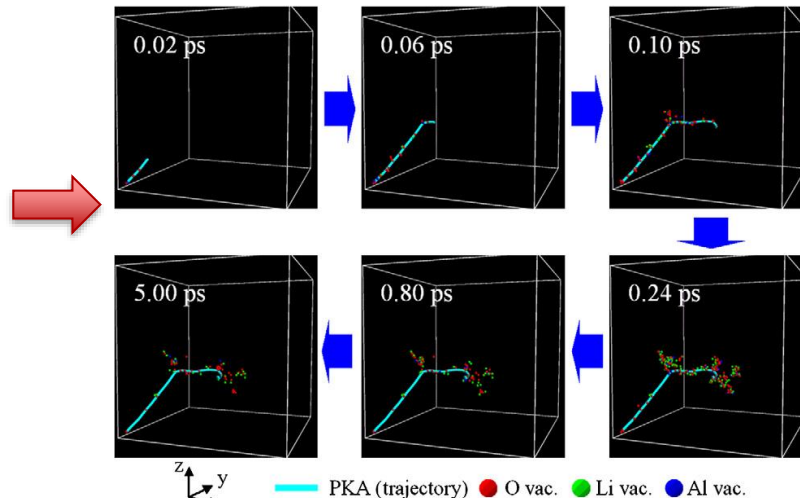
# Bubbles May Affect $^3\text{H}$ Release

- Bubbles form due to He trapping in lattice defects



- Neutron irradiation produces displacement cascades, providing complex defect structures for He or  $^3\text{H}$  to trap in.

MD simulation of displacement cascade in  $\text{LiAlO}_2$  (PKA = 5 keV)

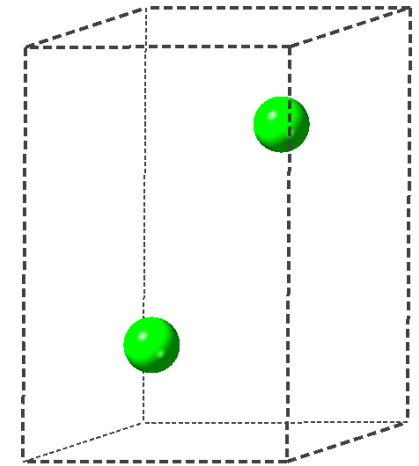


Tsuchihira *et al* JNM 414 (2011) 44-52

# Zircaloy Background

## What is Zircaloy?

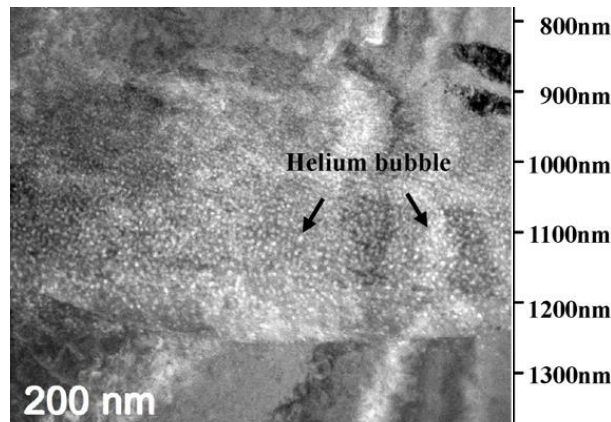
- Zircaloy-2: predominantly used as fuel cladding for BWRs
  - $\alpha$ -Zr, 1.5% Sn, 0.15% Fe, 0.1% Cr, 0.05% Ni
- Zircaloy-4: Removed the Ni and increased Fe content for less H uptake in certain reactor conditions
  - $\alpha$ -Zr, 1.5% Sn, 0.2% Fe, 0.1% Cr
- Zr-Nb alloys (e.g. Zirlo) are also common
- $\alpha$ -Zr has a **hexagonal close-packed (HCP)** crystal structure up to 810°C



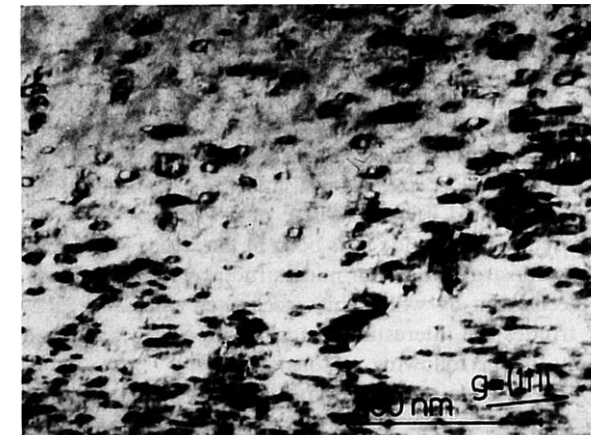
Crystal Structure of  $\alpha$ -Zr (HCP)

## Gas and defect behavior in Zr/Zr alloys

- $^3\text{H}$ , H, and He diffusion and release
- Bubble formation
- Irradiation induced metallic precipitate formation



He bubbles in Zr-Nb alloy  
*Shen et al Mat Char 107 (2015) 309-316*

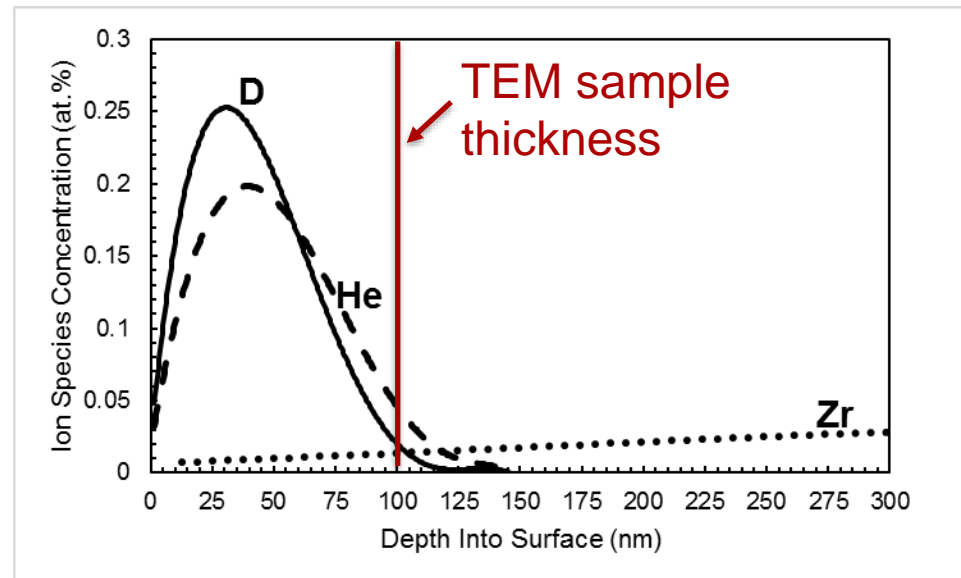


TEM of Zr tritide after 325d  
*Schober et al JNM 141-143 (1986) 453-457*

# Zr alloy in-situ ion irradiation parameters

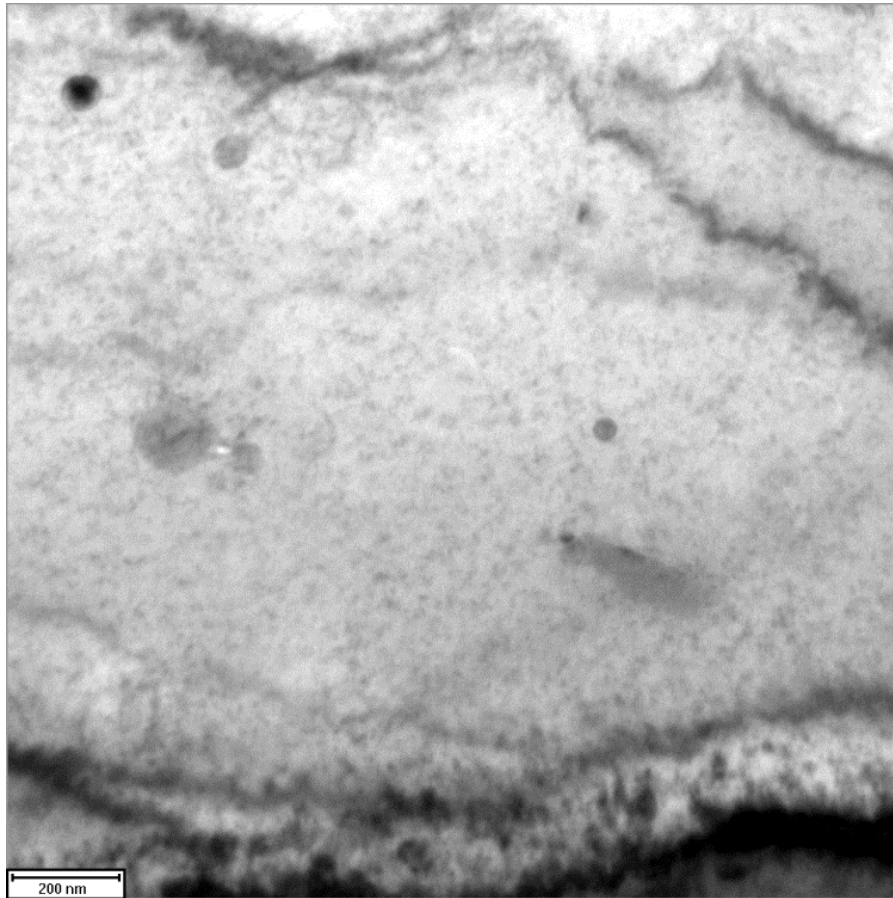
- Samples were prepared by electropolishing zirconium alloy samples (mostly ZIRLO)
- Several sets of irradiations done at 310°C, including:
  - 10 keV He → simulates He accumulation from  $^6\text{Li}$  transmutation and  $^3\text{H}$  decay
  - 10 keV He + 5 keV D + **3 MeV Zr** → simulates gas build-up + displacement cascades
- SRIM, a Monte Carlo based program for simulating the number of displacements produced by an ion, was used to predict damage dose and concentration profiles.
- These preliminary experiments were run overnight and the exact gas concentrations/damage doses are not all known

He and D profiles are implanted within the TEM sample, while most Zr passes through the sample, leaving only cascade damage.

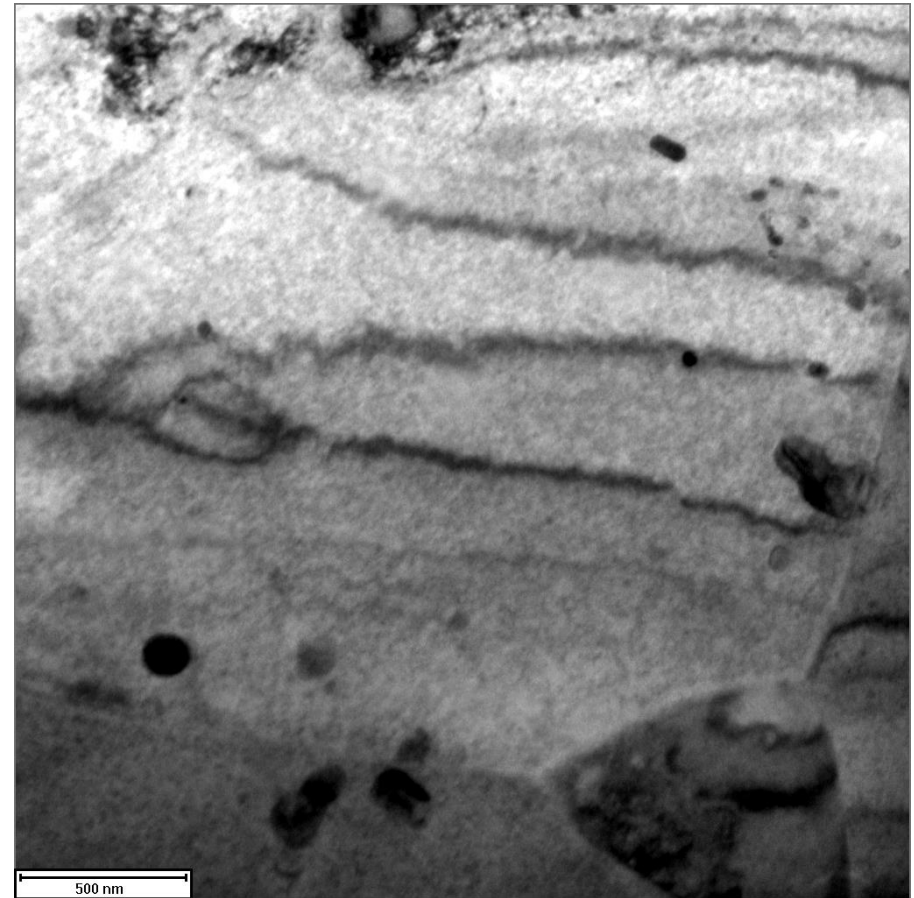




# 10 keV He<sup>+</sup> Implantation at 310°C



After Implantation.  
Damage, No Cavities.



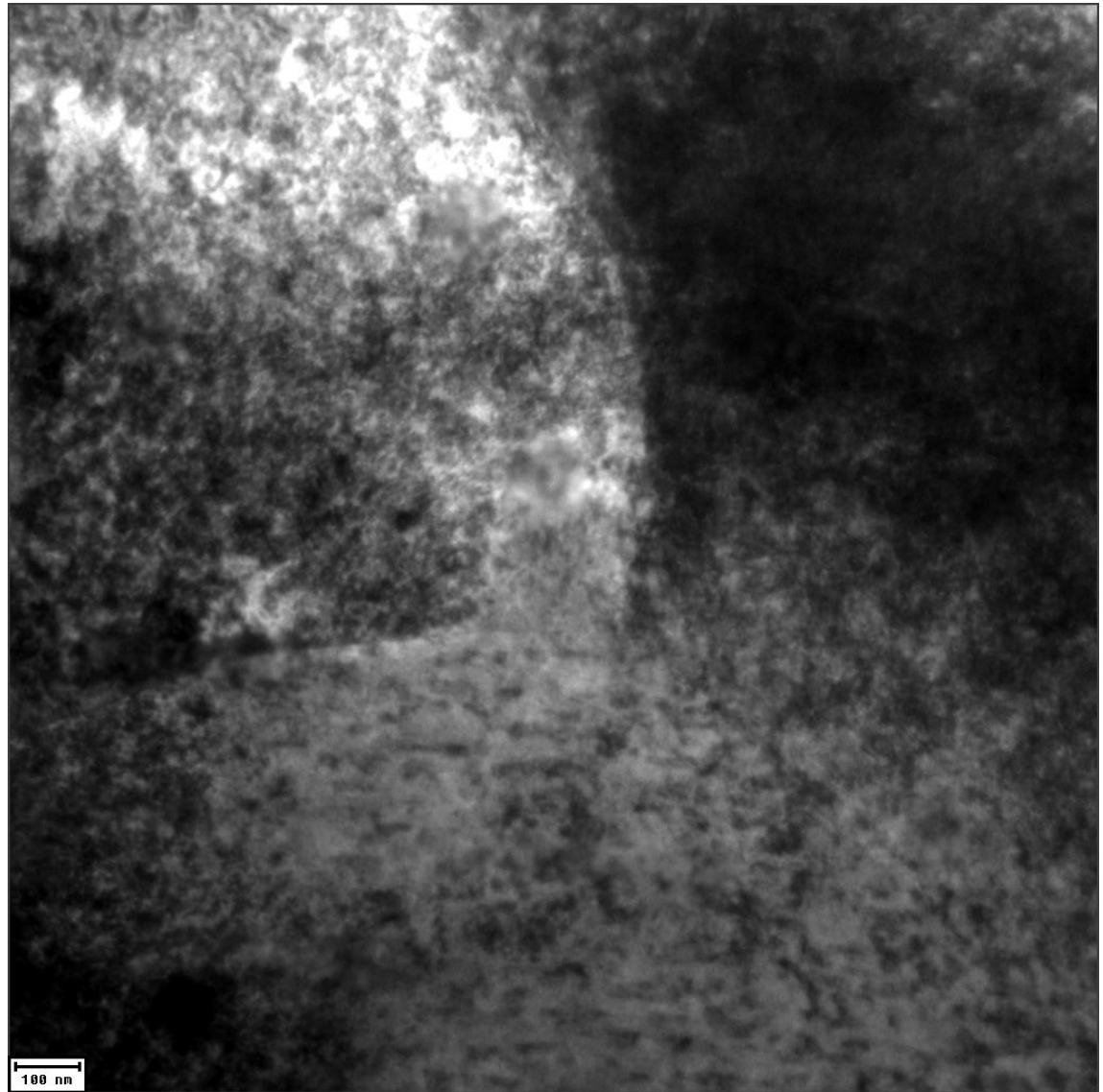
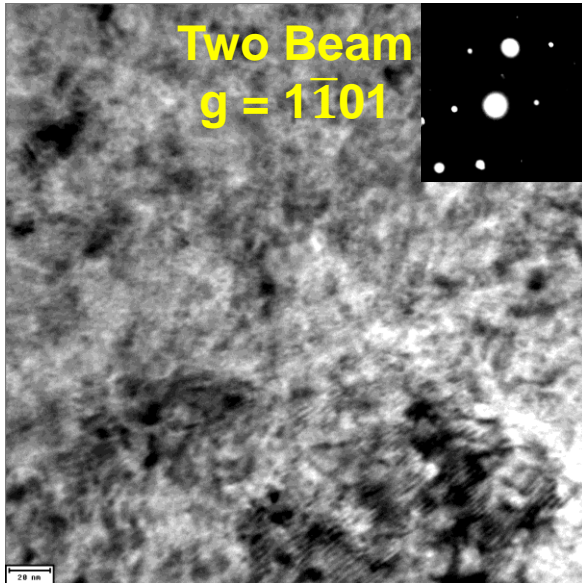
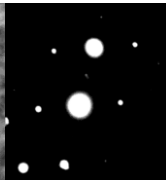
Still no cavities after subsequent  
irradiation with 3 MeV Zr.

# Concurrent D & He Implantation & Zr Irradiation

After triple beam  
irradiation

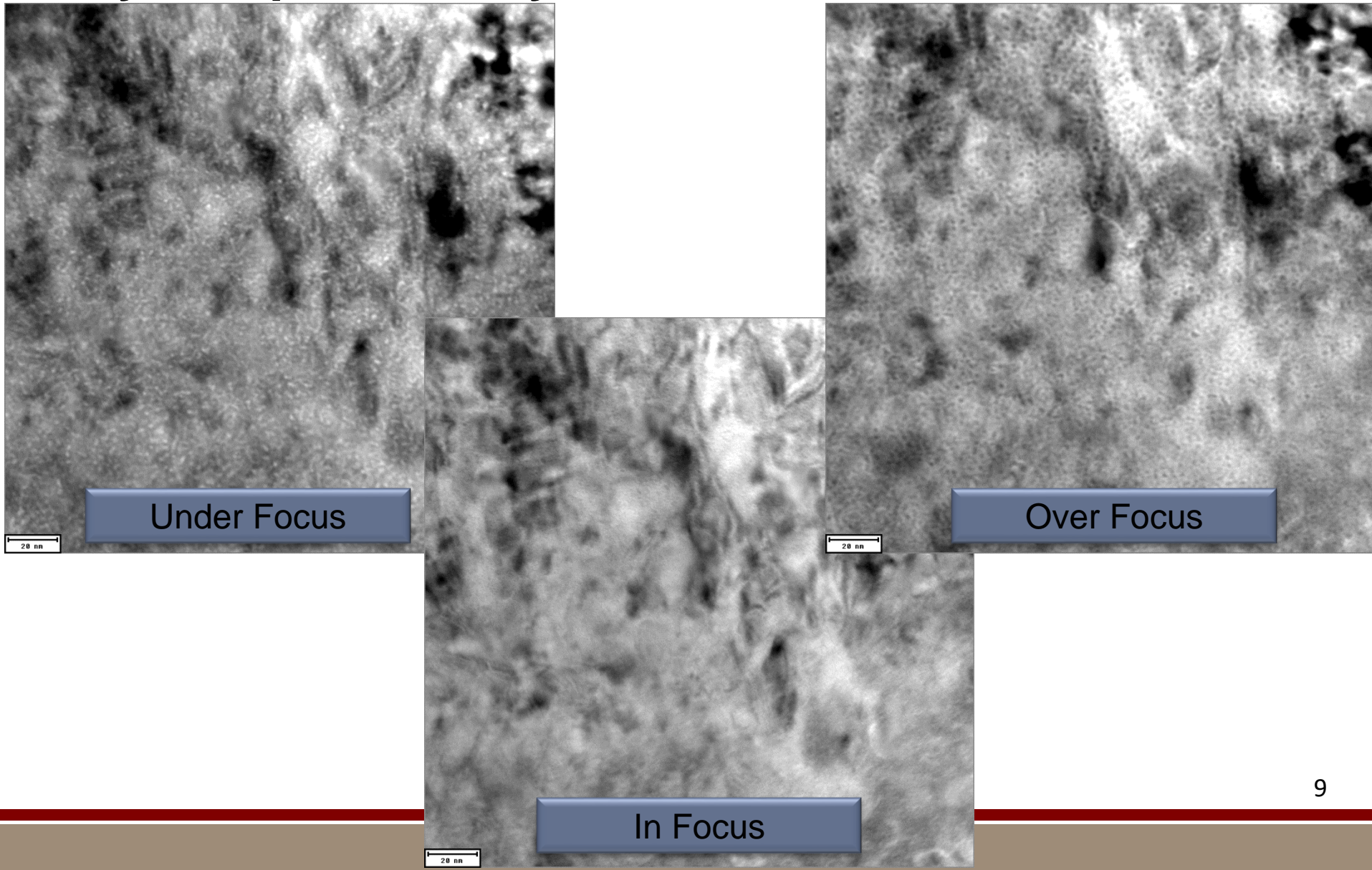
- Very dense, complex defect structure
- No visible cavities
- Fuzzy defects difficult to characterize

Two Beam  
 $g = 1\bar{1}01$

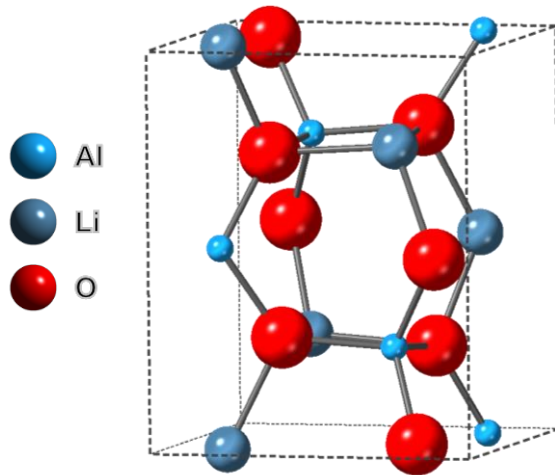




# Cavities were observed in He implanted Zr alloy samples 30 days after irradiation



# LiAlO<sub>2</sub> Background

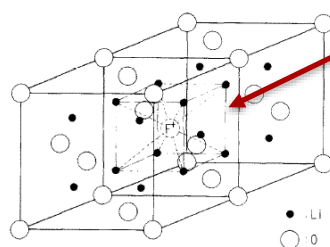


$\gamma$ -LiAlO<sub>2</sub> is tetragonal  
(space group: P 41 21 2)

## Previous Work

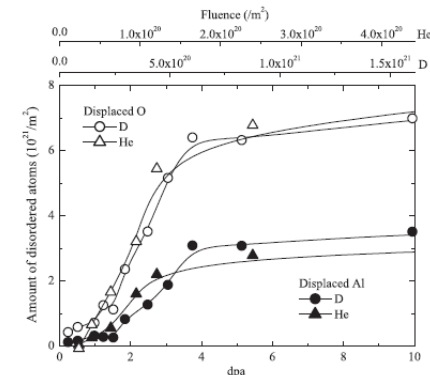
- Structural defects
  - Luo *et al* JNM 372 (2008) 53-58
- Volume swelling
  - Noda JNM 179-181 (1991) 37-41
- <sup>3</sup>H detrapping
  - Oyaidzu *et al* JNM 375 (2008) 1-7
- Gas diffusion and release
  - Raffray *et al* JNM 210 (1994) 143-160

- H isotopes are thought to trap in oxygen vacancies
  - <sup>2</sup>H release occurs at the same temperature as defect annealing in implanted LiAlO<sub>2</sub>



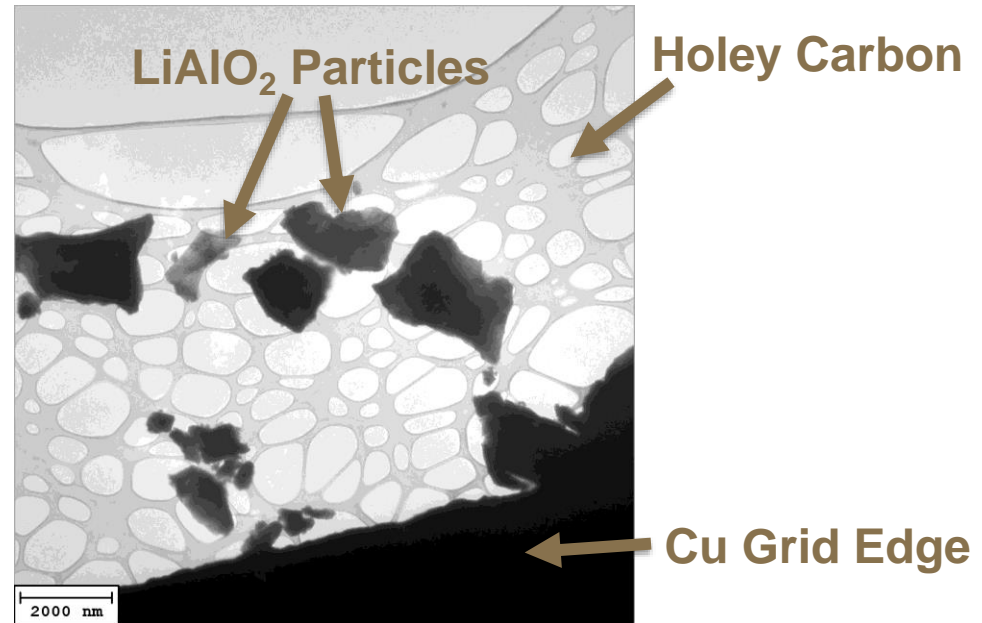
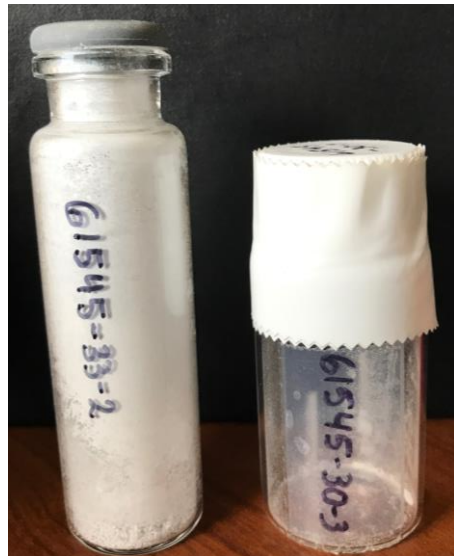
Can be determined  
with luminescence

Noda JNM 179-181 (1991) 37-41



# LiAlO<sub>2</sub> in-situ ion irradiation parameters

- Powders were drop-cast onto TEM grids

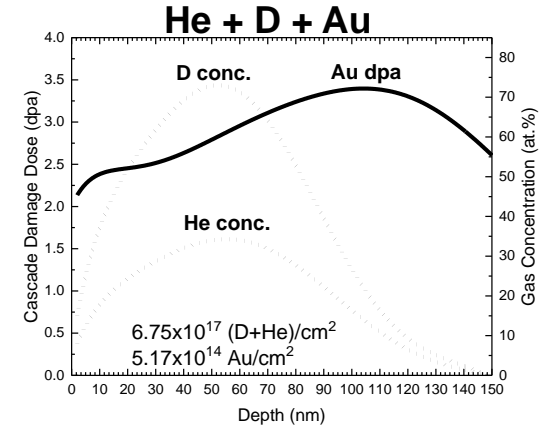
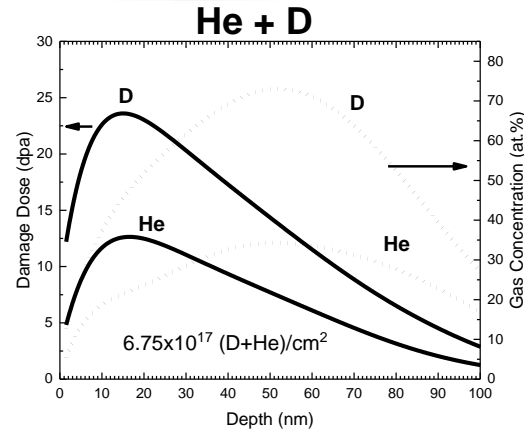
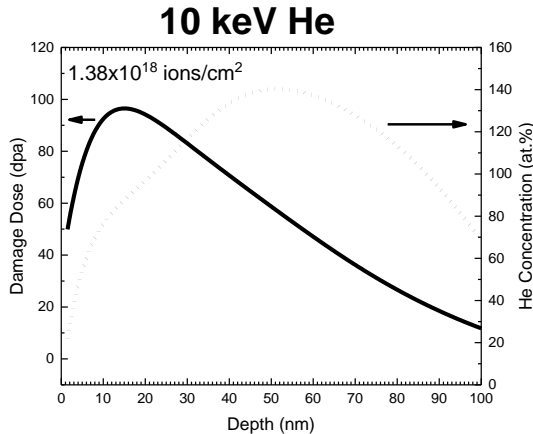


- Samples were heated to **310°C** using Hummingbird HT stage
- Three sets of irradiations:
  - 10 keV He → **simulates He accumulation from <sup>6</sup>Li transmutation and <sup>3</sup>H decay**
  - 10 keV He + 5 keV D → **simulates He and <sup>3</sup>H interaction**
  - 10 keV He + 5 keV D + **1.7 MeV Au** → **simulates gas build-up + displacement cascades**

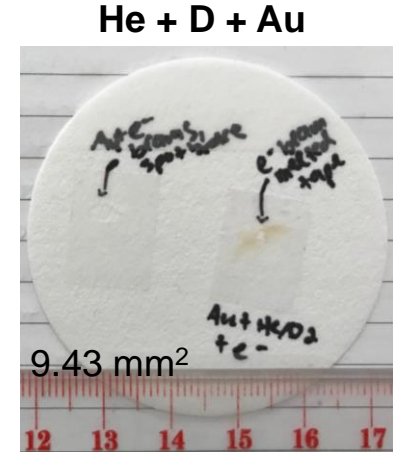
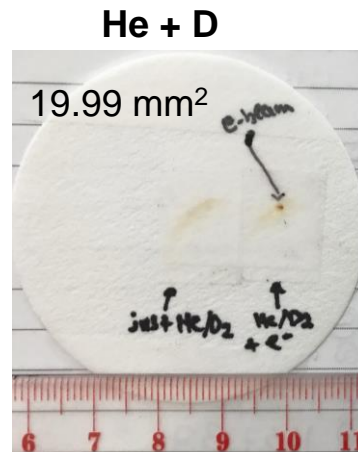
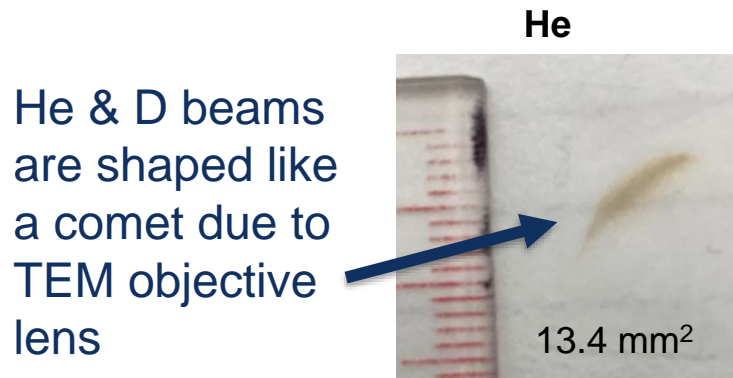
# LiAlO<sub>2</sub> in-situ ion irradiation parameters

## SRIM Calculations

Most He/D diffuses out of thin film immediately



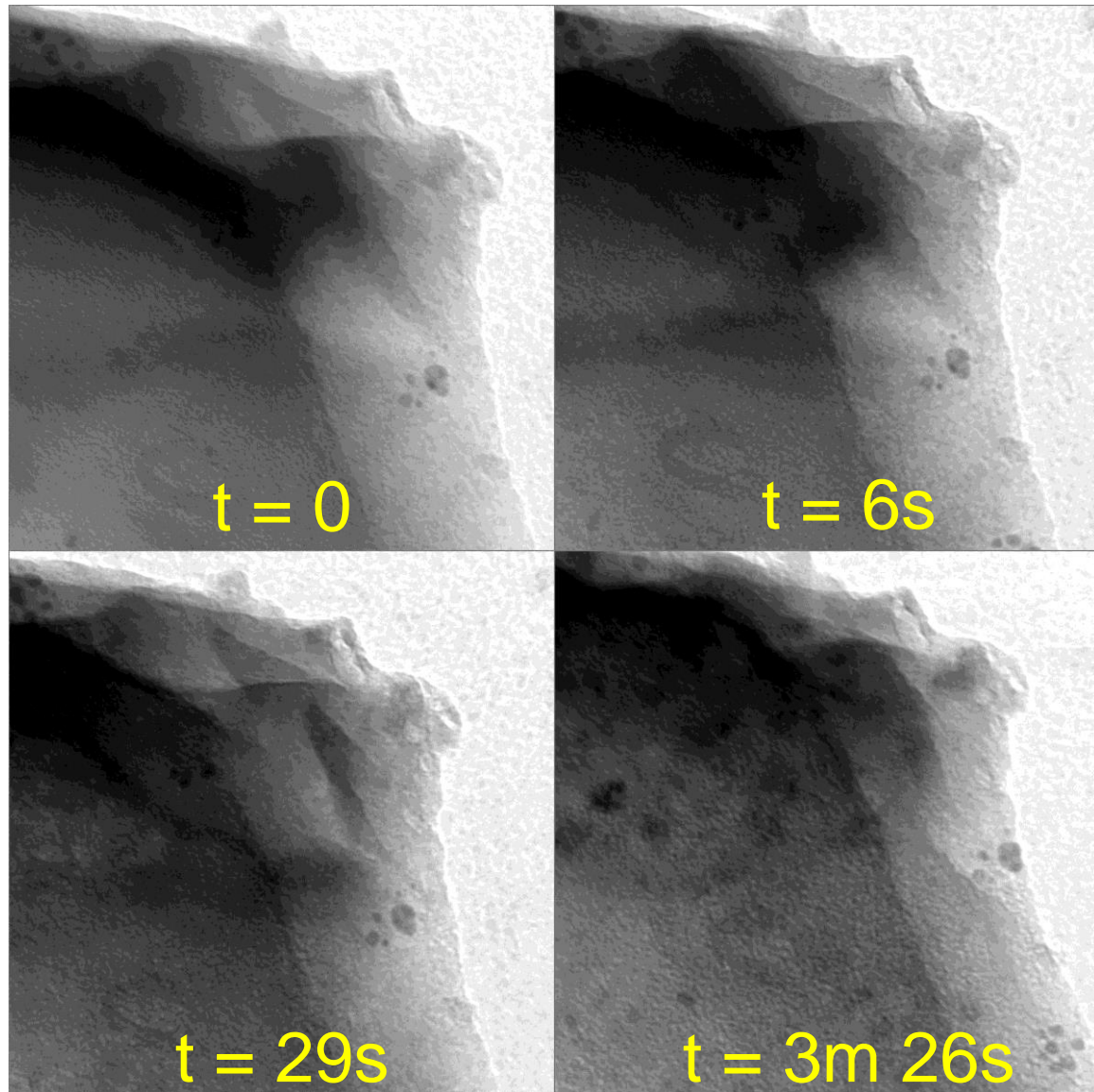
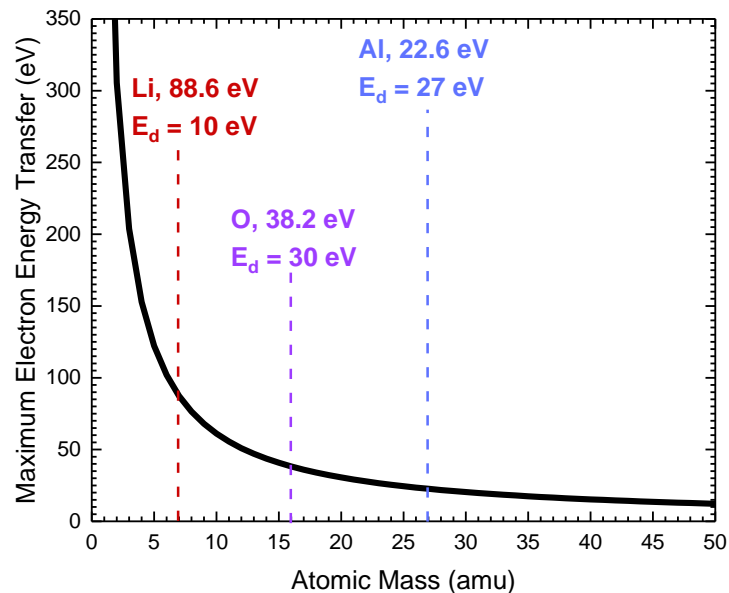
- Burn spots were used (1) to confirm that electron beam spot overlaps ion beam spot and (2) to determine the ion beam irradiation area.





# Electron beam induced void growth

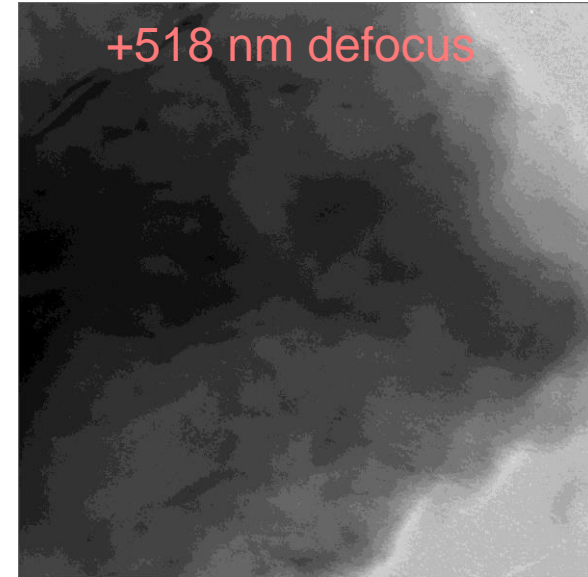
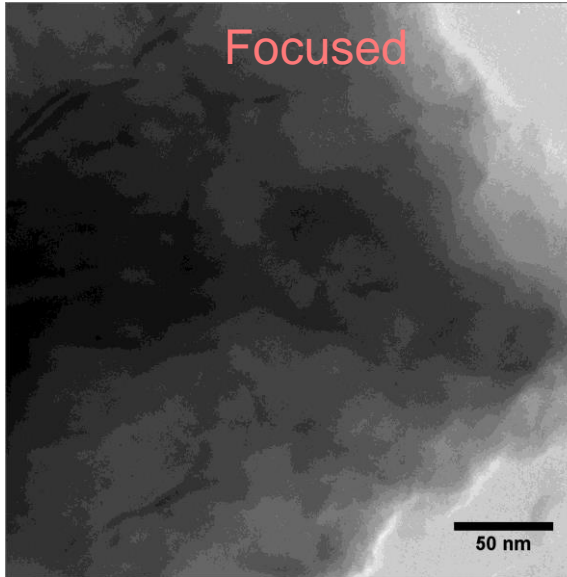
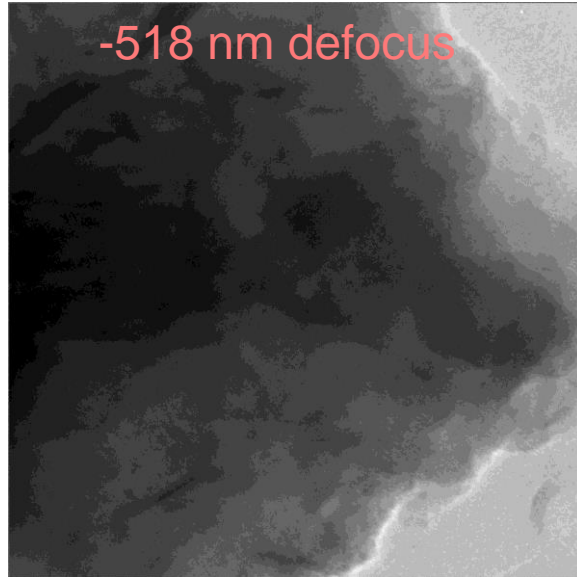
- Voids were observed to form under the electron beam in several particles
- Rate of void formation is not consistent between particles
- Possibly due to electron beam displacing Li and O atoms



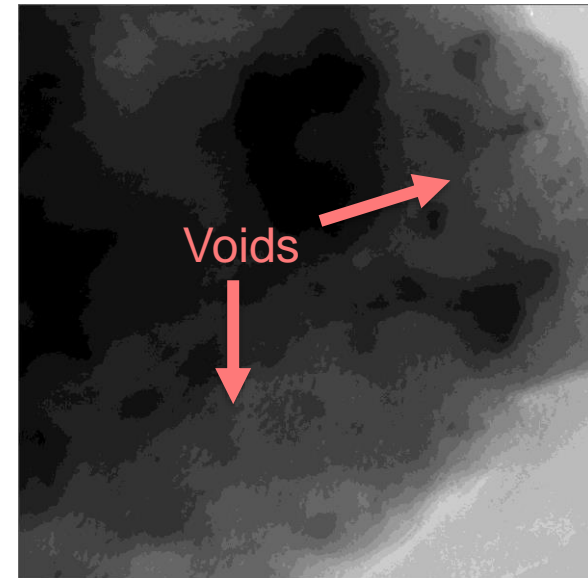
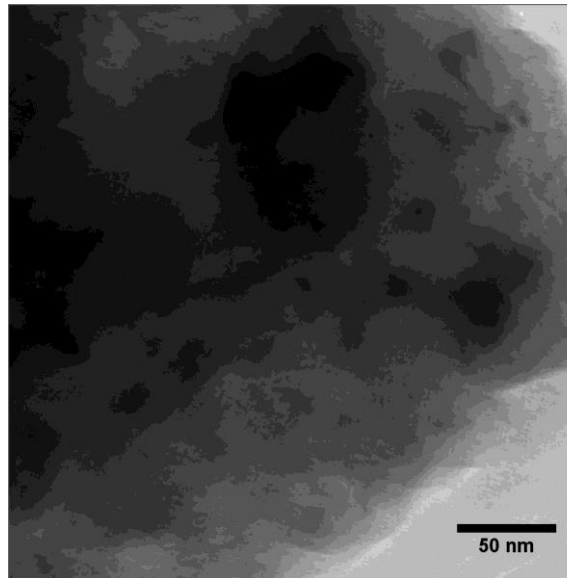
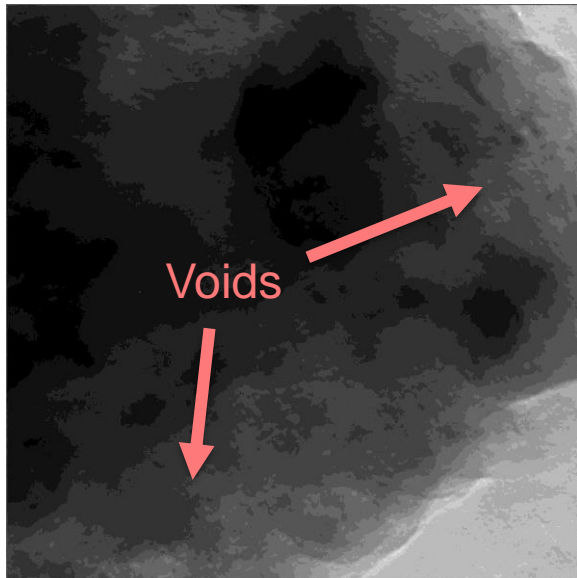


# In-situ He implantation @ 310°C

Before



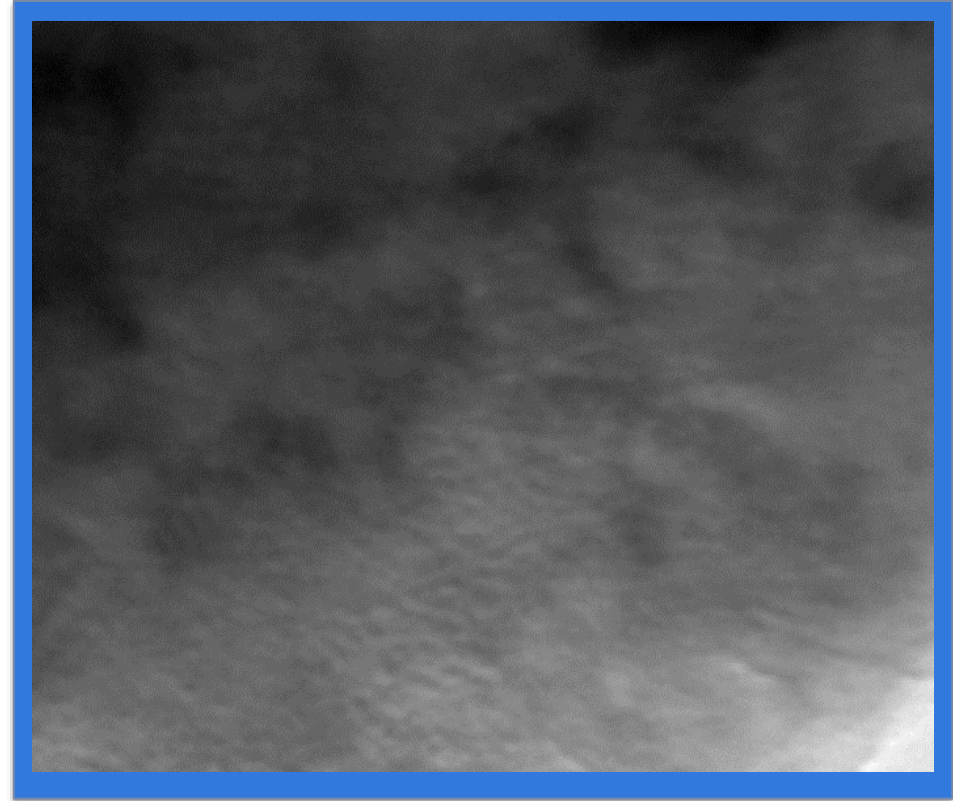
After



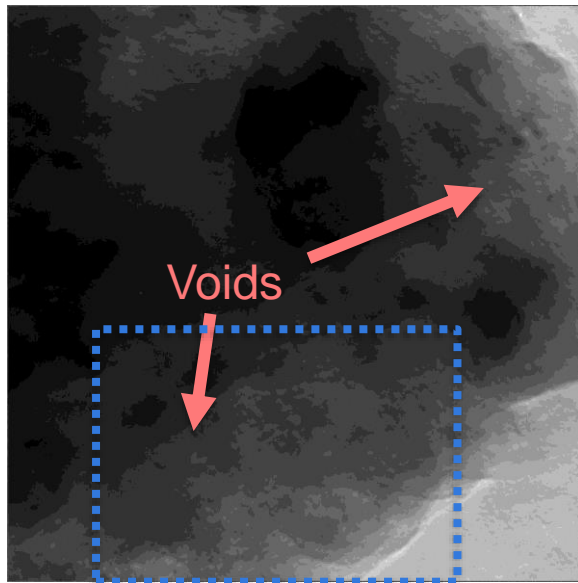
# In-situ He implantation @ 310°C

- Each frame = 1 min of irradiation
- Because the voids are difficult to see in powders, I paused the video at a few points to show overfocus images
- Electron beam on for most of experiment
- Bubbles formed after ~13 min ( $1.5 \times 10^{17}$  He/cm<sup>2</sup>)

## In-situ Video

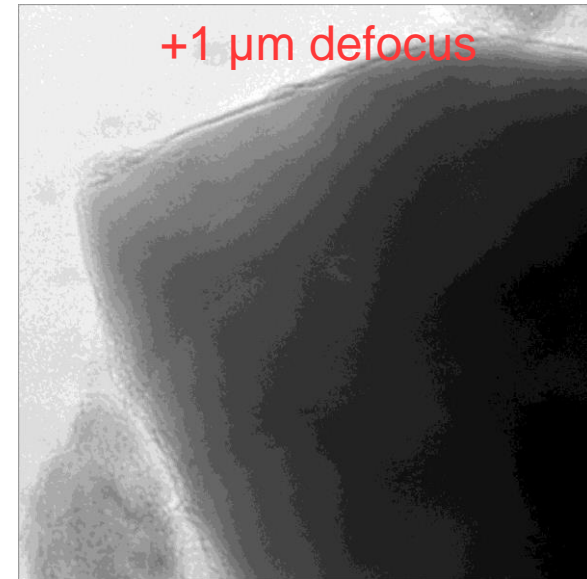
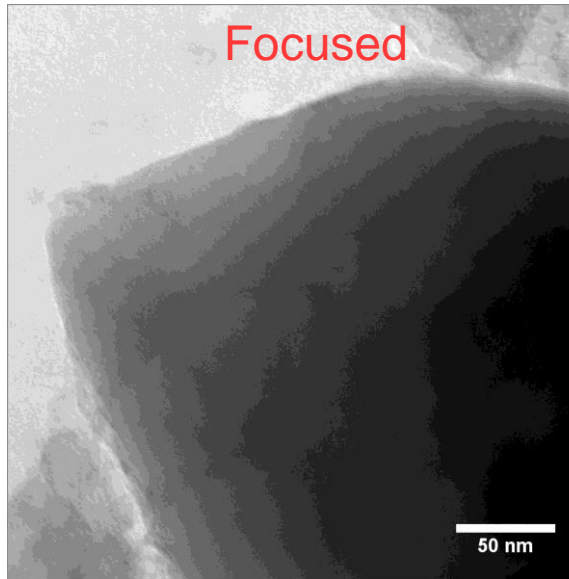
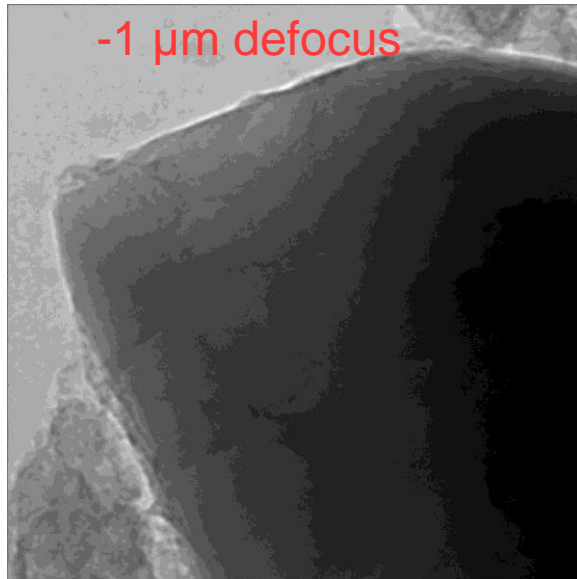


## After Irradiation

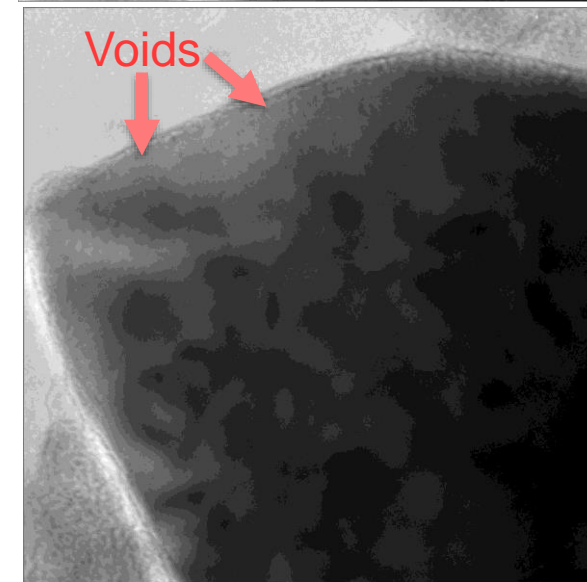
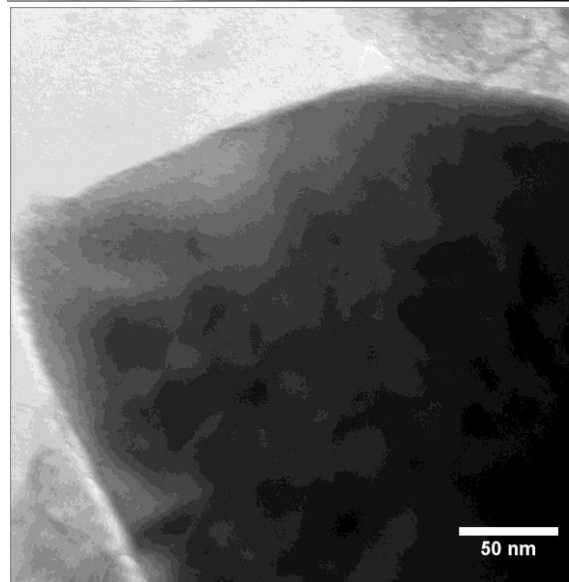
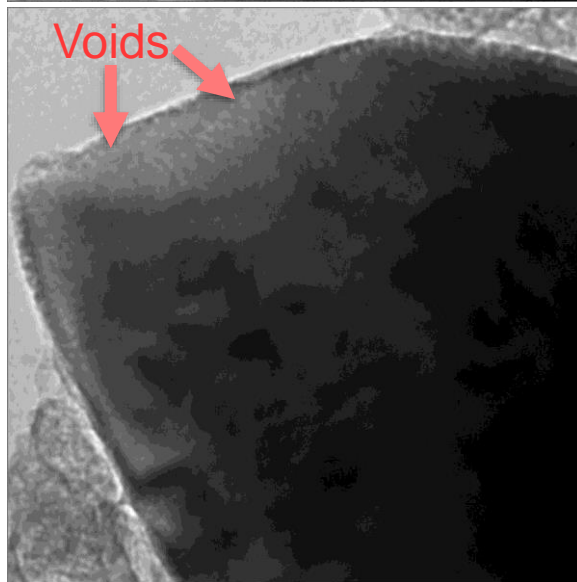


# In-situ He + D irradiation @ 310°C

Before



After

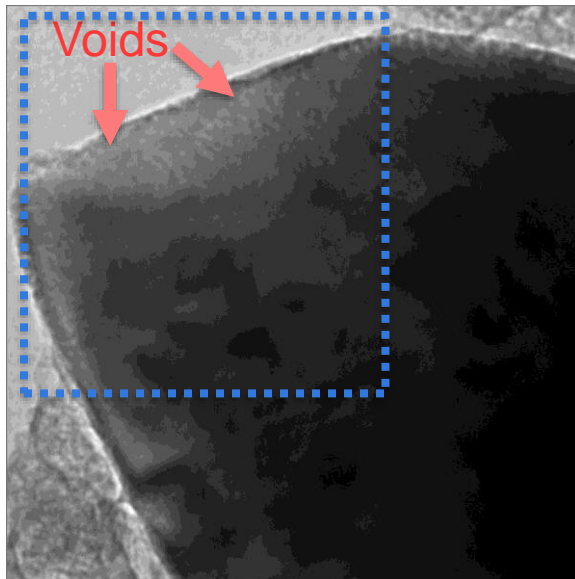




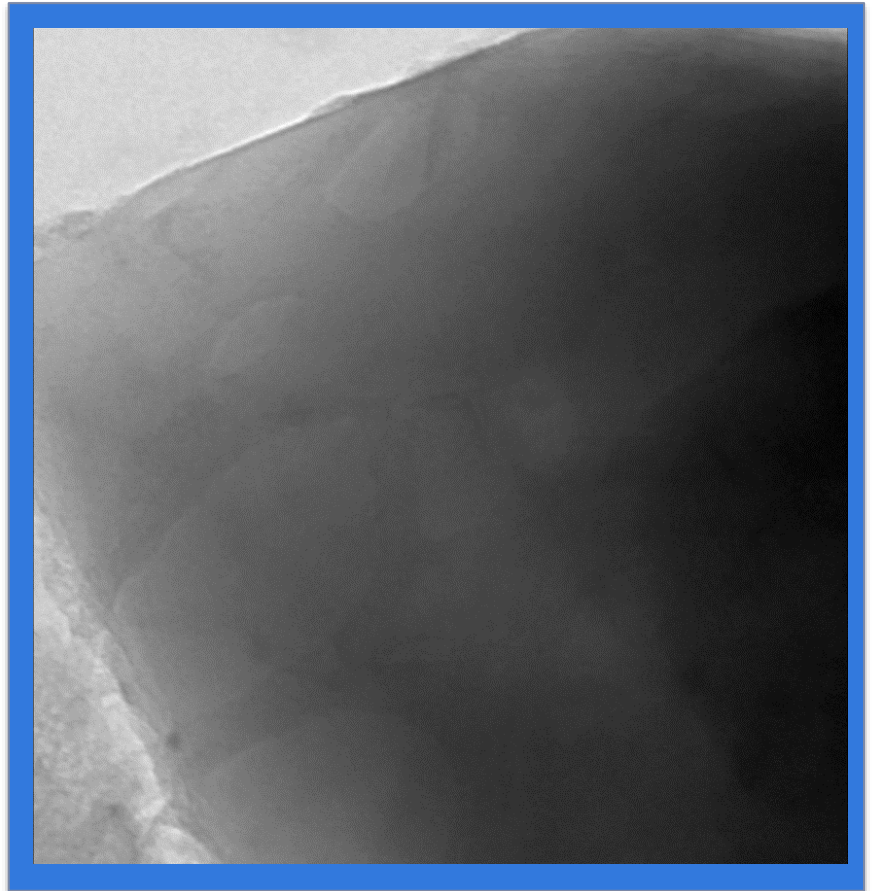
# In-situ He + D irradiation @ 310°C

- Each frame = 5 min of irradiation
- All underfocus images
- Electron beam was off except for imaging
- Bubbles formed after ~60 min ( $1.7 \times 10^{17}$  He/cm<sup>2</sup>,  $3.4 \times 10^{17}$  D/cm<sup>2</sup>)

## After Irradiation

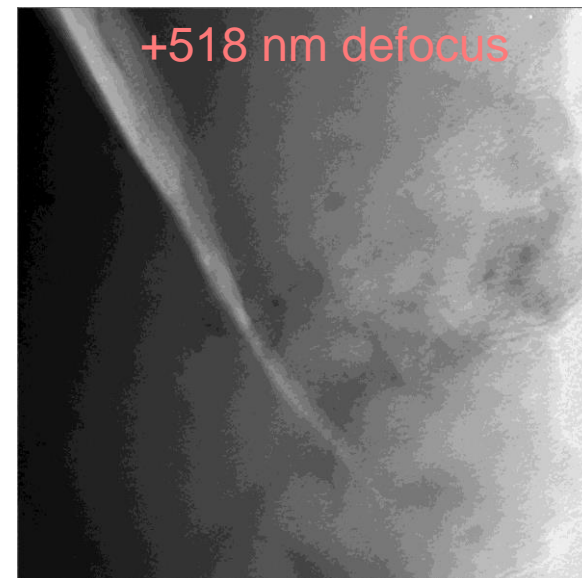
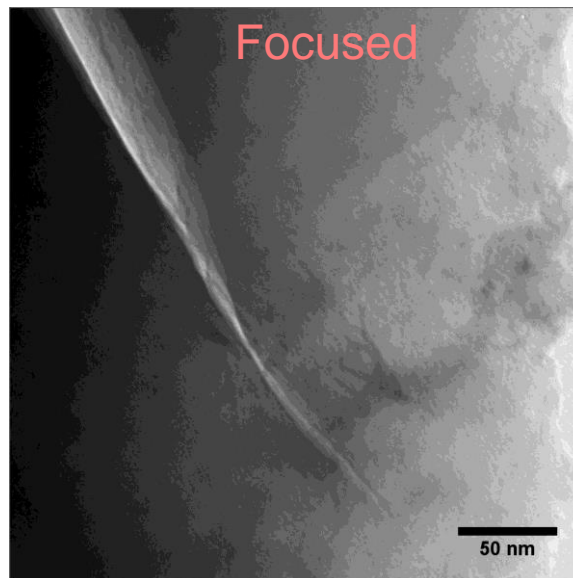
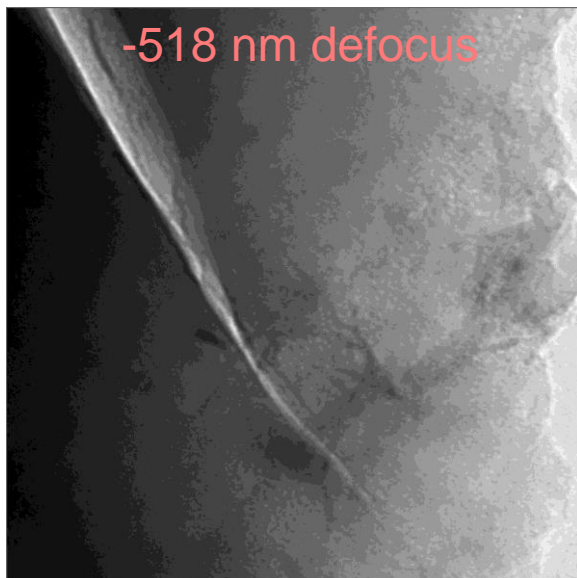


## In-situ Video

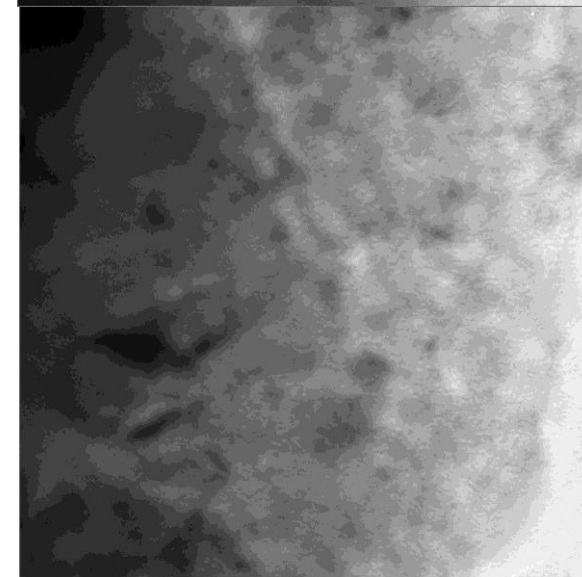
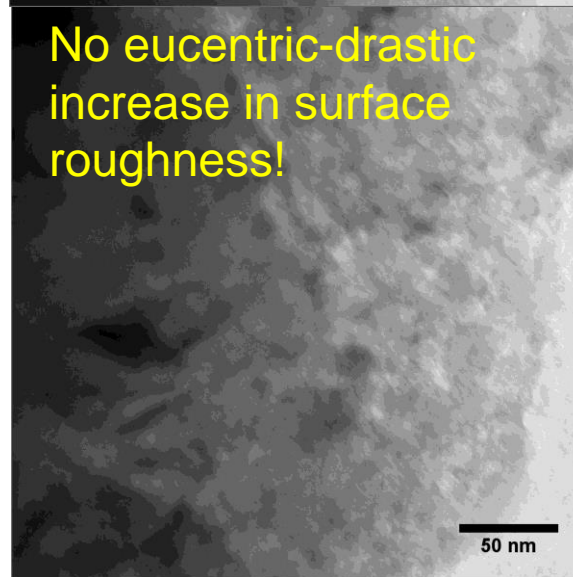
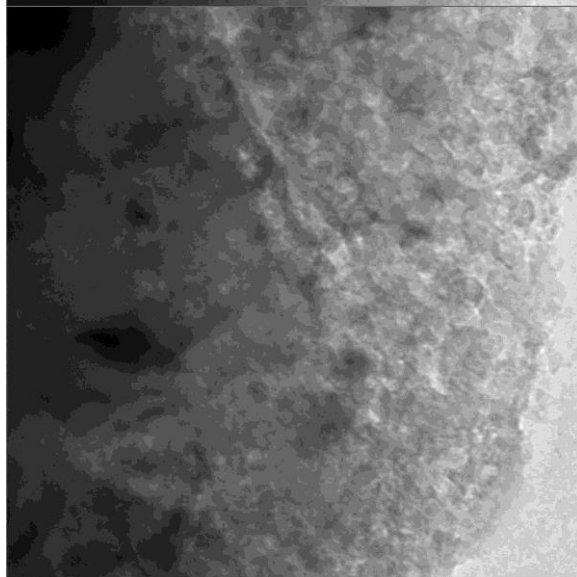


# In-situ He + D + Au @ 310°C

Before



After

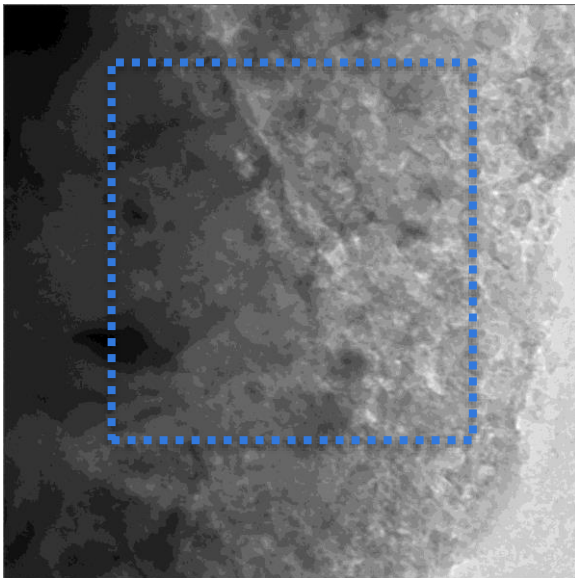




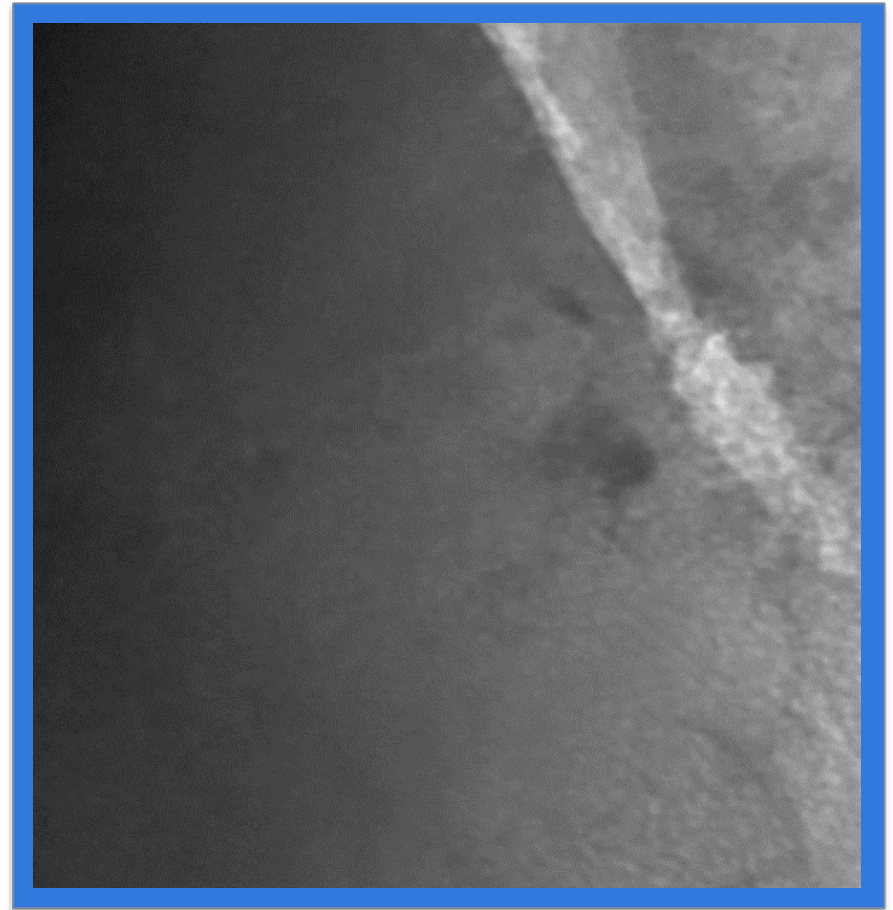
# In-situ He + D + Au @ 310°C

- Each frame = 5 min of irradiation
- Pre-existing voids could have an effect on this final microstructure
- Electron beam was on for most of the experiment

**After Irradiation**



**In-situ Video**



# Summary

Zr Alloys

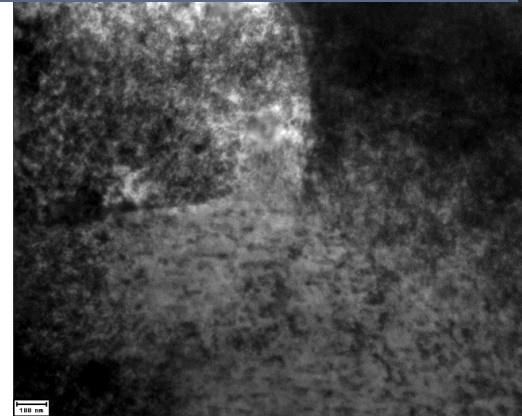
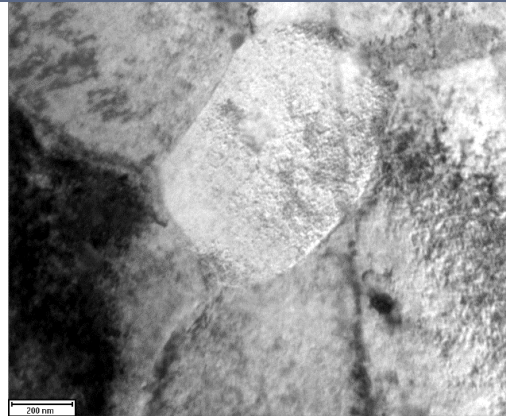
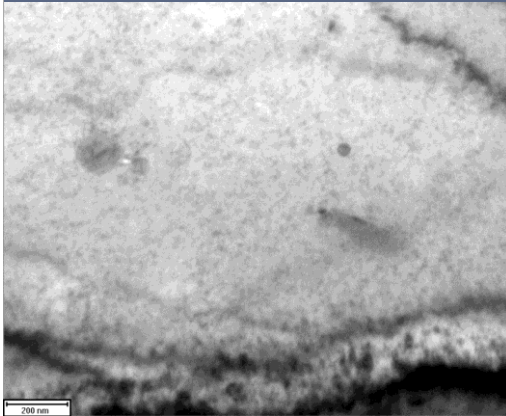
He accumulation



Damage + He



Damage + He +  $^3\text{H}$



LiAlO<sub>2</sub>

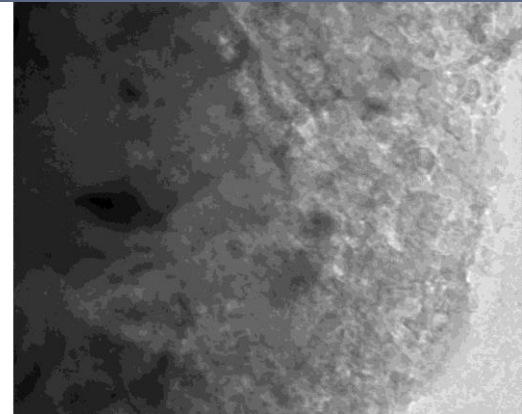
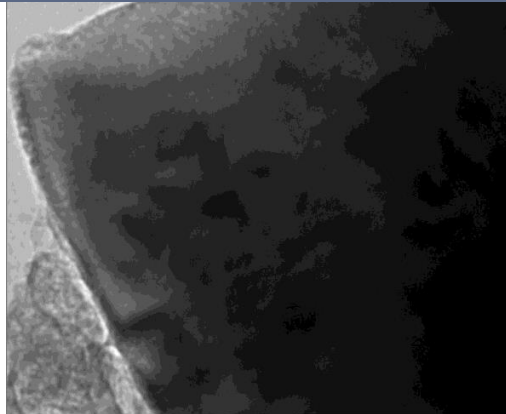
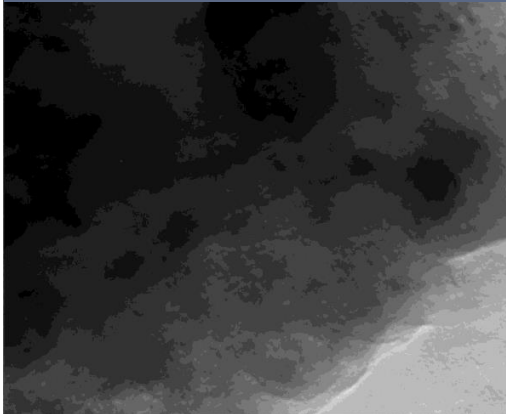
He accumulation



$^3\text{H}$  + He



Damage + He +  $^3\text{H}$



- Synergistic effects between He and  $^3\text{H}$  accumulation and neutron irradiation induced displacement cascades were simulated in-situ at reactor temperature.
- Microstructure resulting from triple ion irradiation is complex in both materials studied, and will require additional studies to understand defect gas interactions in this complex environment.