

# *Correlating Microstructural Quality with Thermoelectric Properties to Optimize $Bi_{1-x}Sb_x$ Thin Films*

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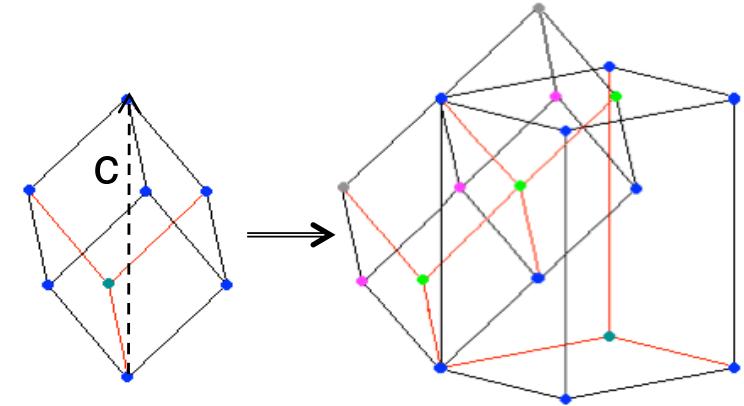
# Outline

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- 1. Introduction to  $\text{Bi}_{1-x}\text{Sb}_x$**
- 2.  $\text{Bi}_{0.8}\text{Sb}_{0.2}$  film growth**
  - Deposition and post-annealing experiments
  - Role of SiN protective cap
- 3. Microstructural characterizations (XRD, SEM)**
- 4. Transport properties (resistivity, Seebeck)**
- 5. Structure-property correlations**
- 6. Summary**

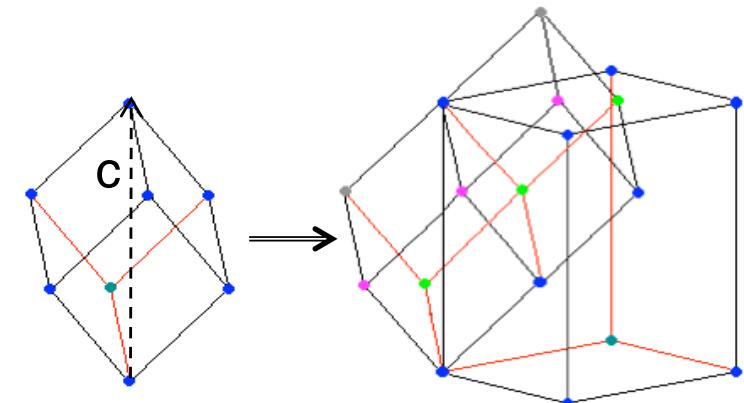
# Introduction to $\text{Bi}_{1-x}\text{Sb}_x$

- Individually, Bi and Sb are semimetals that form continuous  $\text{Bi}_{1-x}\text{Sb}_x$  solutions.
- Bulk  $\text{Bi}_{1-x}\text{Sb}_x$  is semiconducting for  $0.06 < x < 0.22$  with optimal TE properties between 70 – 100K.
- Rhombohedral structure creates an anisotropy in electrical properties that are optimal along the trigonal axis.

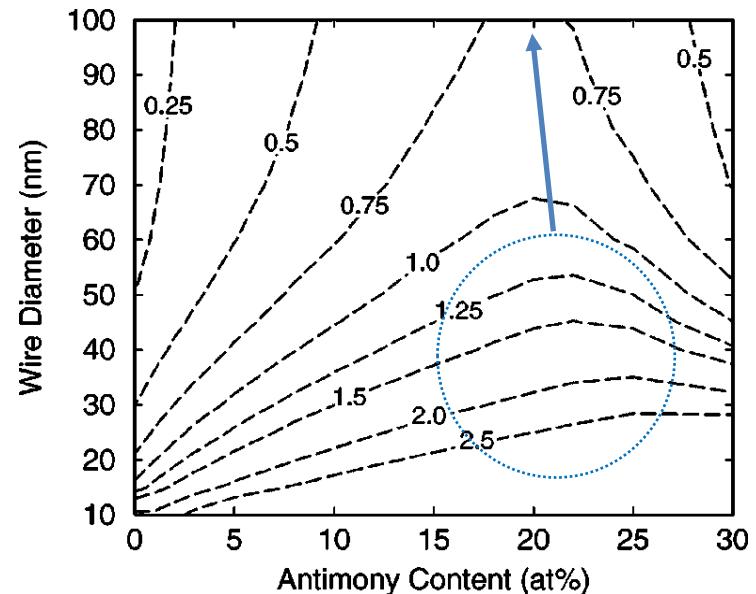


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- Rhombohedral structure creates an anisotropy in electrical properties that are optimal along the trigonal axis.
- Single crystal and polycrystalline bulk studies find maximum ZT near  $x = 0.1$ .
- Quantum confinement calculations predict higher ZT for  $0.20 < x < 0.25$ .
- Since we studied 100 nm thick films, we selected  $x = 0.20$  ( $\text{Bi}_{0.8}\text{Sb}_{0.2}$ ).



## ZT (77K) for nano-sized $\text{Bi}_{1-x}\text{Sb}_x$



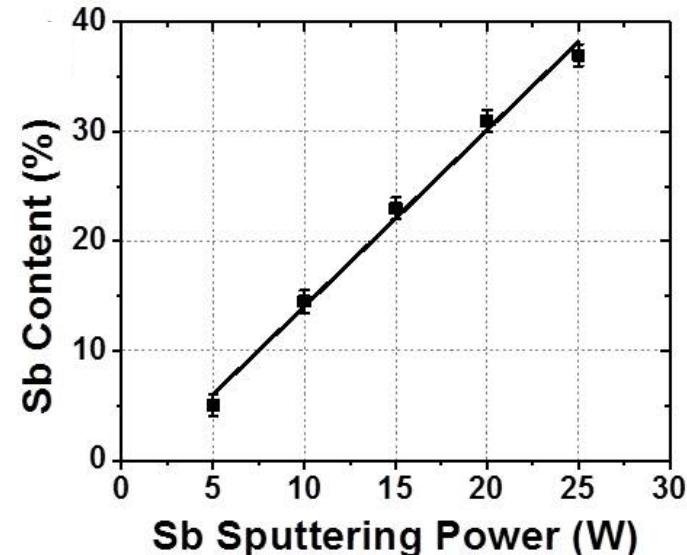


# $Bi_{1-x}Sb_x$ Film Deposition

*Co-deposit Bi and Sb using RF-sputter deposition for compositional control.*

## Control over film composition:

- RBS measurements to  $\pm 1\%$ .
- Bi, Sb co-sputtered on rotating substrate for  $\sim 2"$  diameter uniformity.
- Bi sputtering power held constant at 25 W.
- Sb sputtering power varied.
- Sputtering pressure of 10 mTorr chosen for smoothest films.

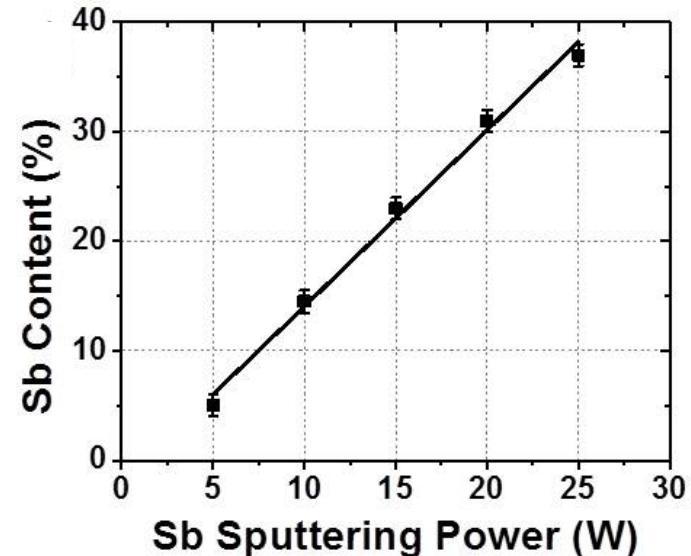


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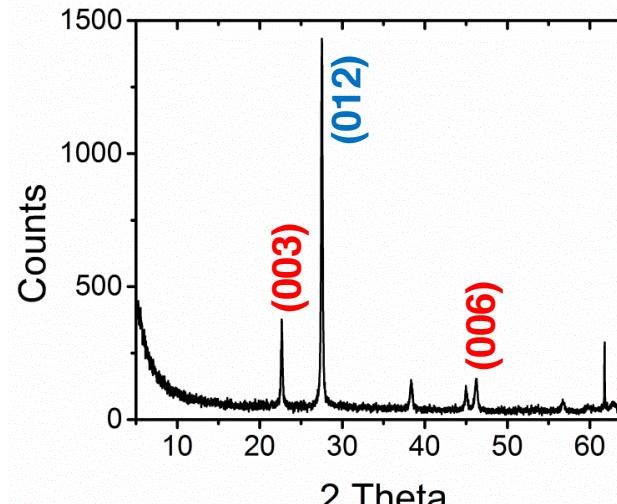
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## As-deposited BiSb films:

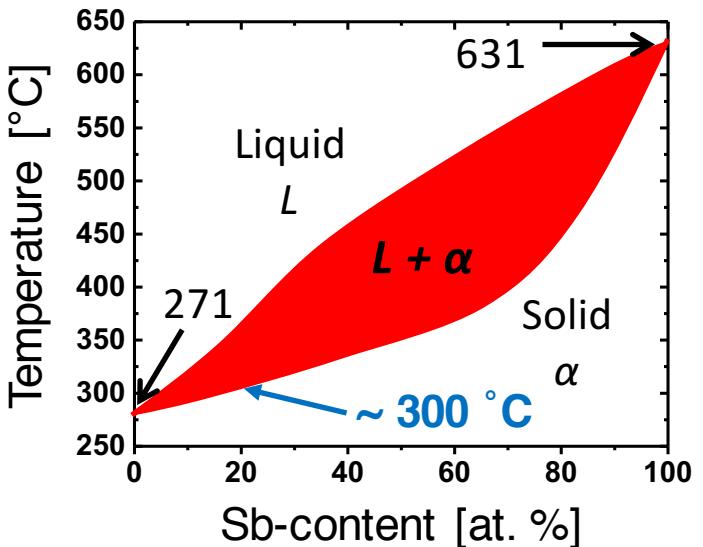
Exhibit weak diffraction peaks with NO preferred orientation, *essentially similar to a powder sample*.



# $\text{Bi}_{0.8}\text{Sb}_{0.2}$ *ex situ* Annealing

- All films annealed in 3%  $\text{H}_2$  / 97%  $\text{N}_2$  forming gas to minimize oxidation.
- Rapid cooling is needed to maintain trigonal orientation, **suggesting that this is not the lowest free energy surface.** [Limmer et al, *Nano Lett.* (2014)]
- For **20% Sb alloys**, it is critical to remain below **300 °C** to prevent sample melting.

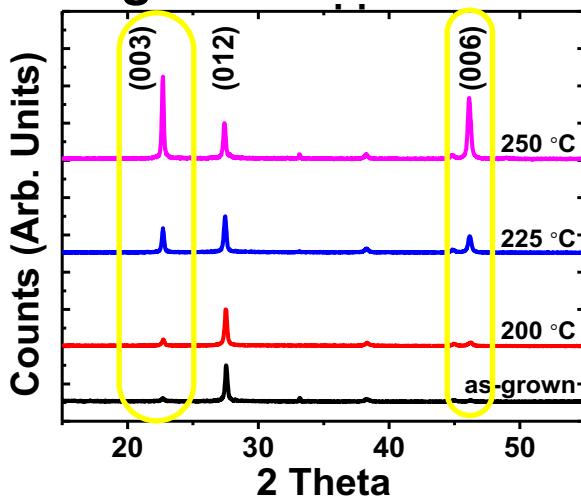
BiSb Alloy Phase Diagram



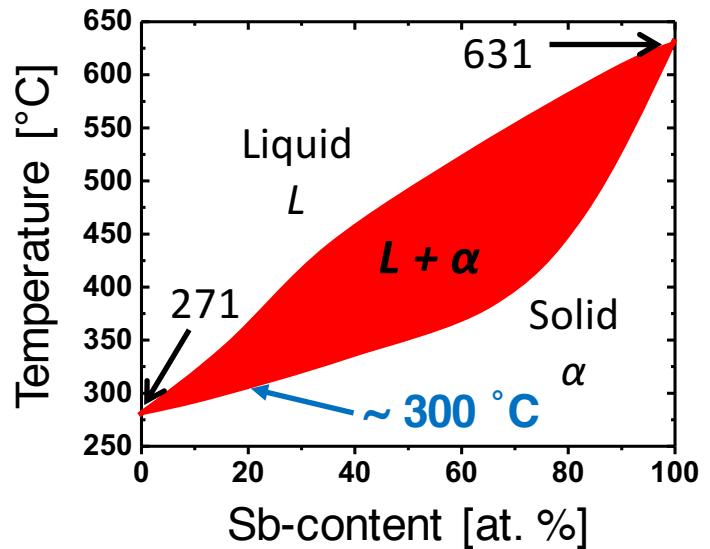
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**Annealing improves trigonal orientation!**



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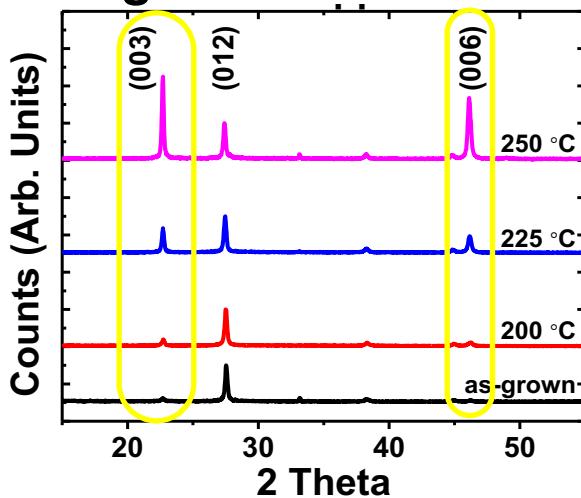


[Rochford et al, *APL Mater.* (2015)]

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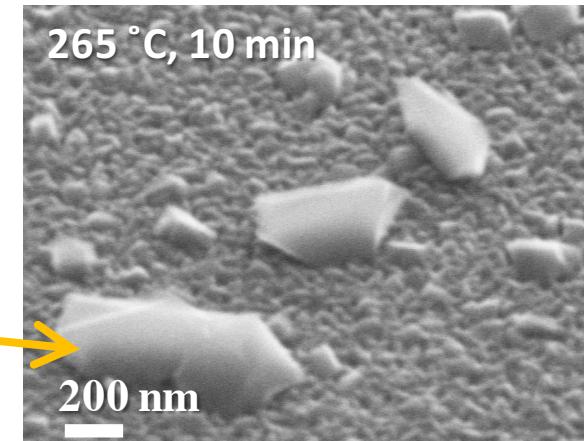
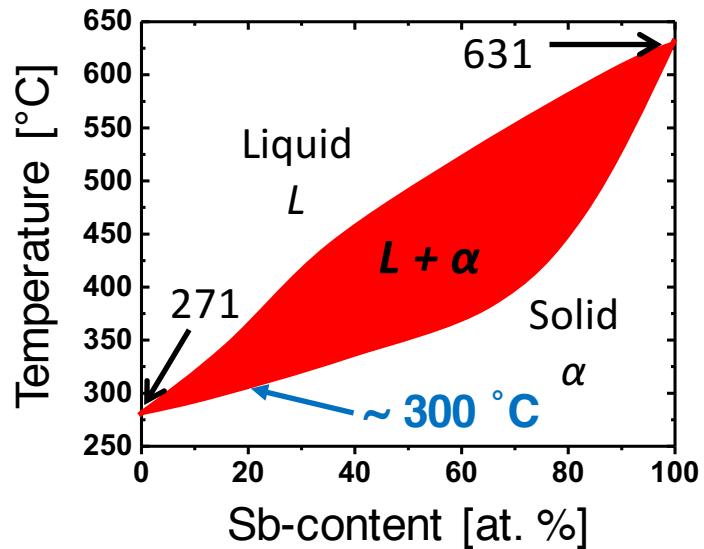
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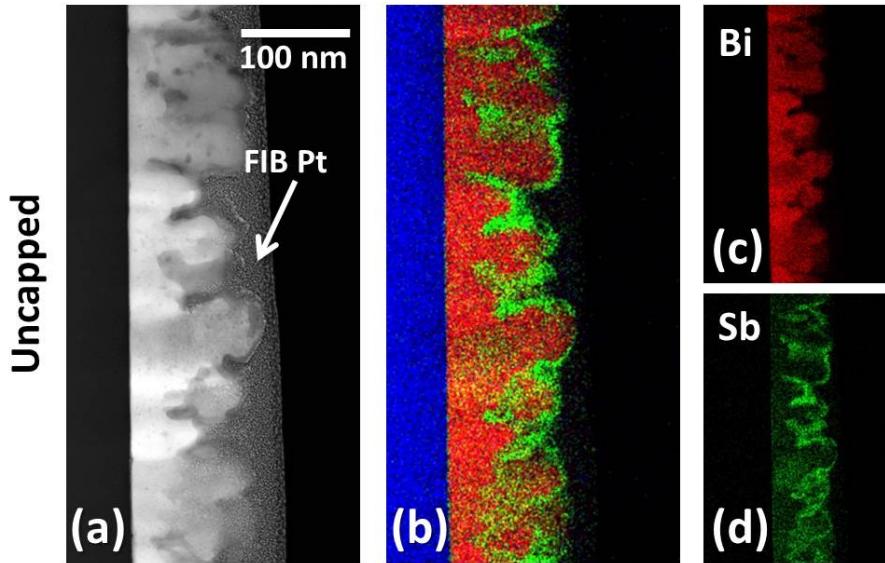
- However, surface roughens with large grain formation.
- Selected area diffraction in TEM indexes large surface grains as  $\text{Sb}_2\text{O}_3$ .
- Where does O come from?

## BiSb Alloy Phase Diagram



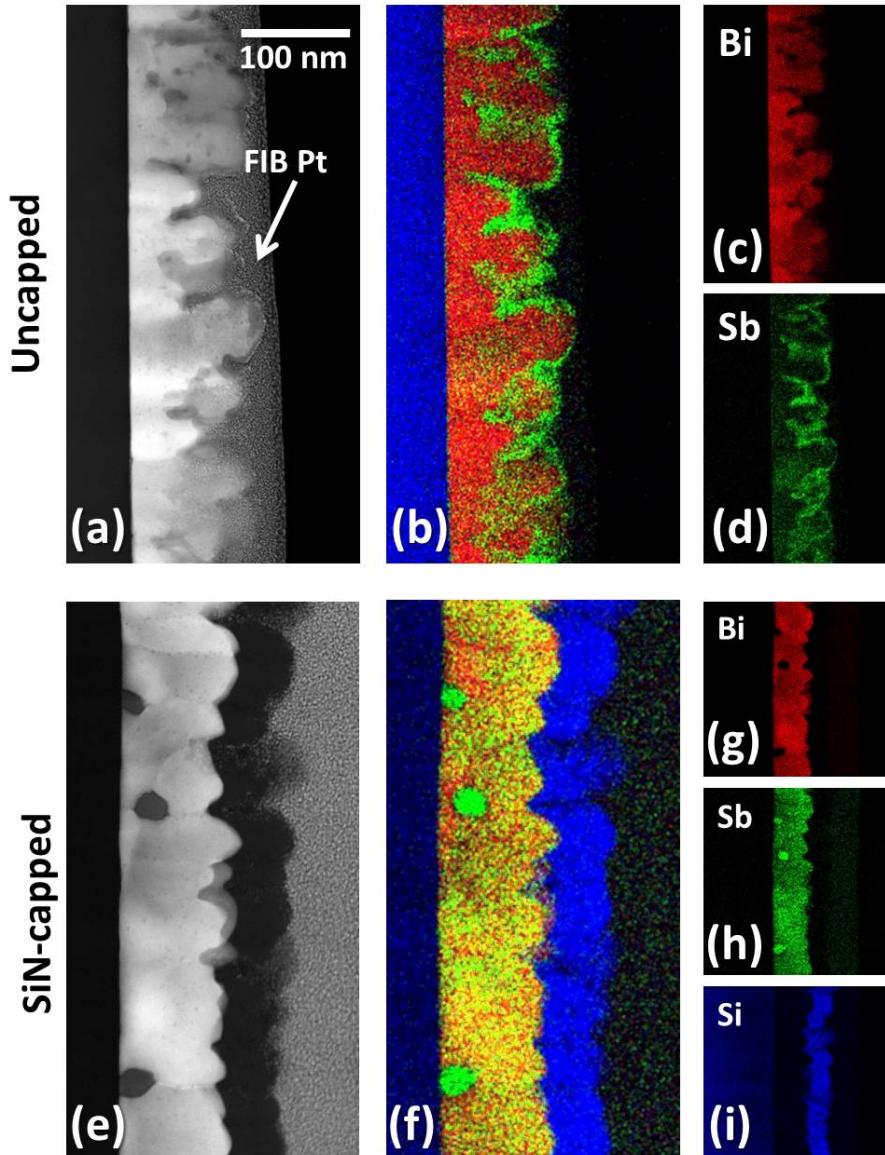
[Rochford et al, *APL Mater.* (2015)]

# Role of Surface Oxide and SiN Protective Capping Layers



- Since films annealed in forming gas, oxide layer must form immediately on as-deposited film upon exposure to air.
- Further analysis with cross-sectional TEM and EDS finds that without capping, **between 50 – 95 % of the Sb segregates to the surface.**
- Hence, such films **WILL NOT** have the assumed composition from deposition.  
***They will be Bi-rich.***

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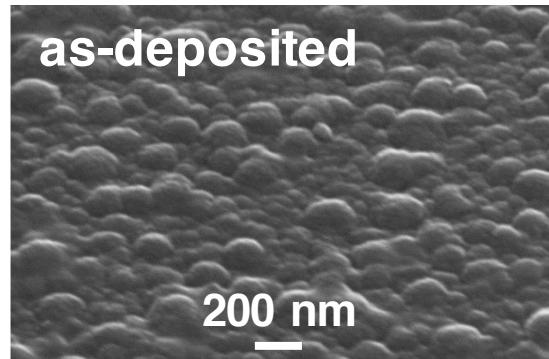


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- Hence, such films **WILL NOT** have the assumed composition from deposition.  
***They will be Bi-rich.***
- **50 nm thick SiN conformal cap layer** prevents oxide formation and results in a smoother films with uniform Bi-Sb composition throughout the film, even after annealing.
- **Should result in films with desired TE properties for a given composition.**

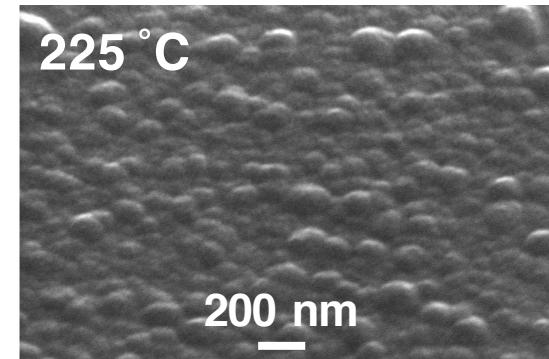
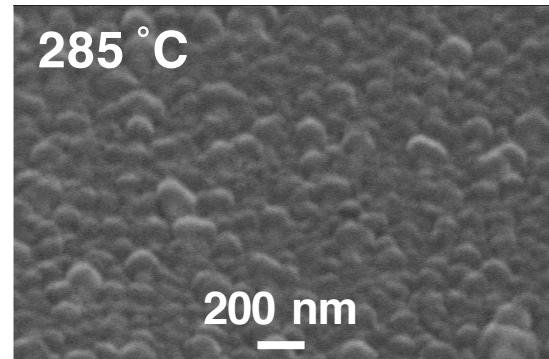
# $\text{Bi}_{0.8}\text{Sb}_{0.2}$ Film Morphology vs. Annealing

*How does annealing SiN-capped films affect morphology?*

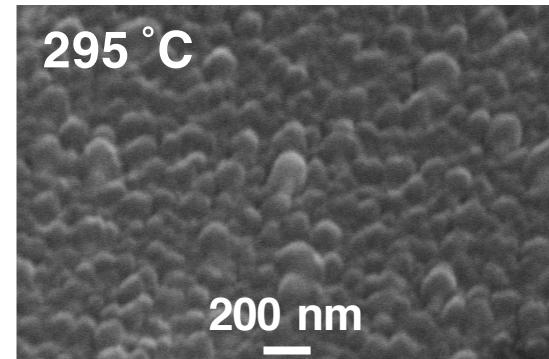
rough morphology  
and large grain  
size variation



best morphology  
and most uniform  
grain sizes



annealing smooths  
morphology and  
homogenizes grain  
sizes

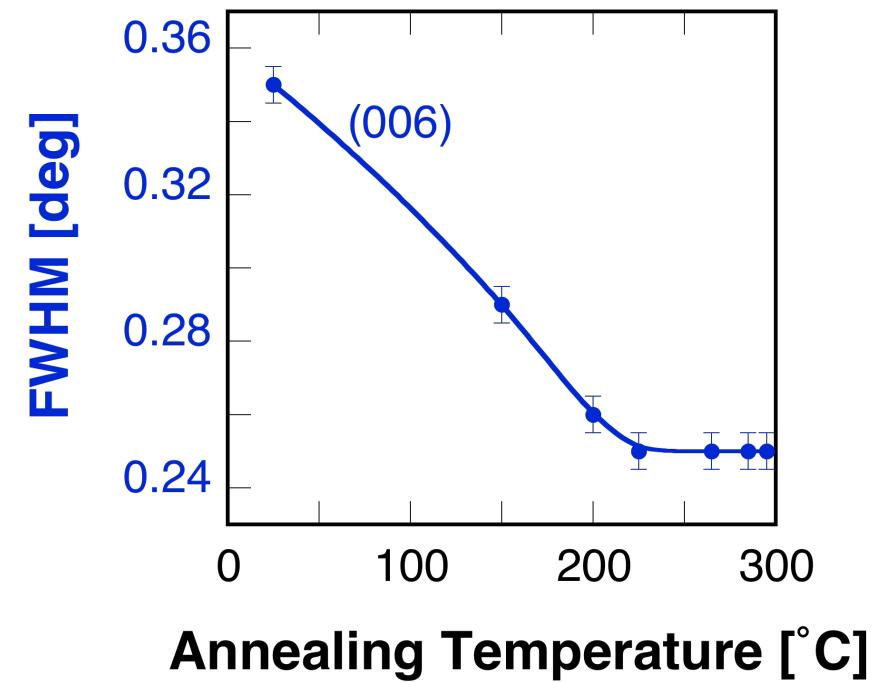
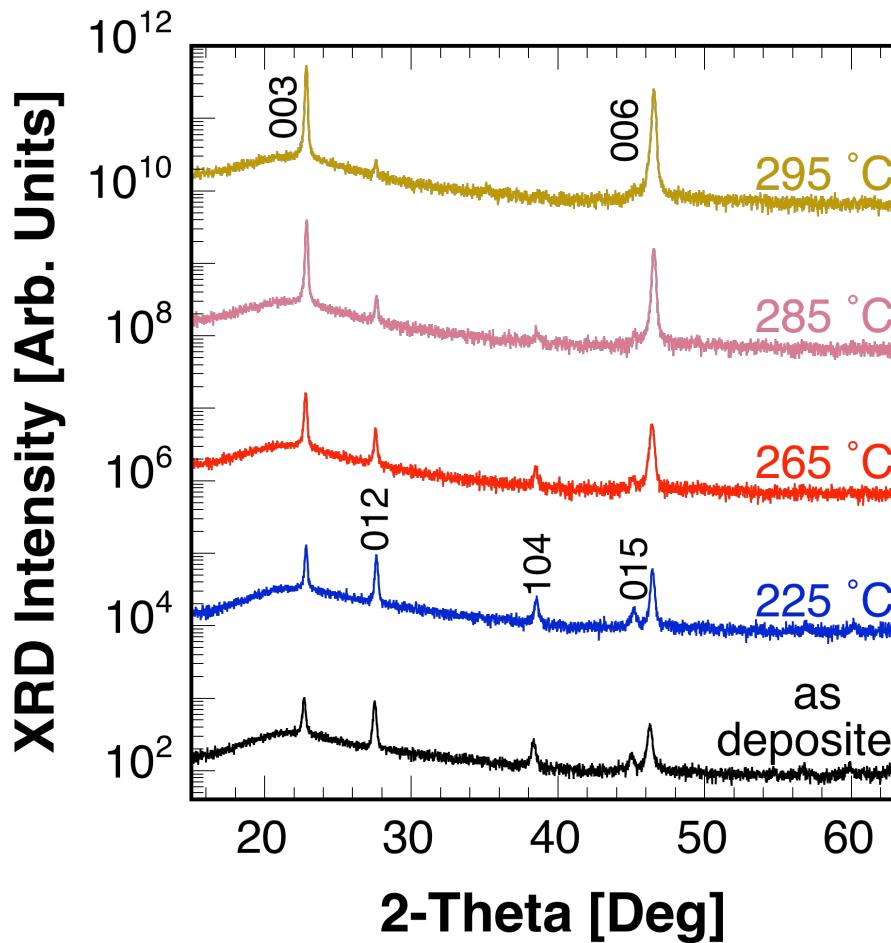


very close to  
melting point,  
distinct grain  
boundaries.

*Annealing leads to smoother morphologies and more homogeneous grain sizes, however, grain boundaries become more distinct close to the melting point.*

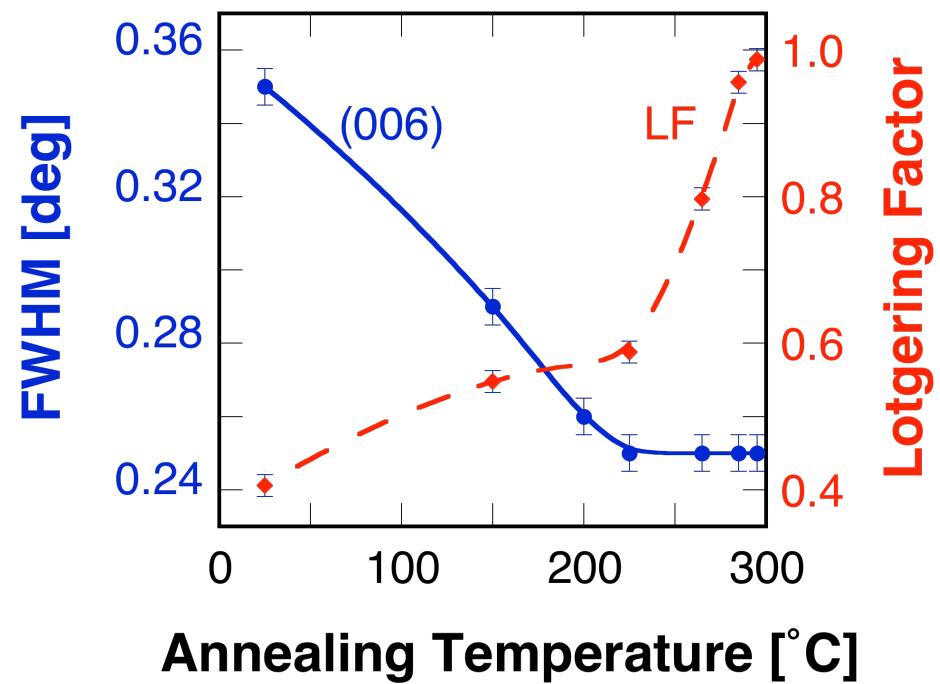
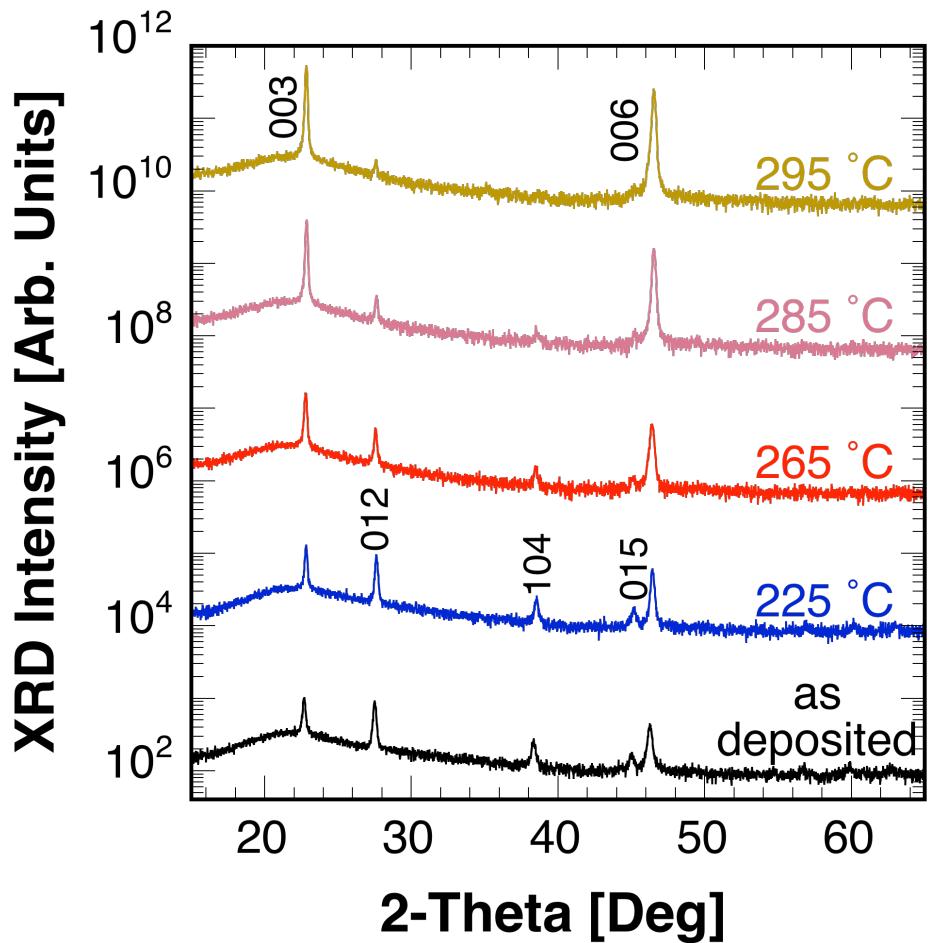
# Annealing to Further Improve $\text{Bi}_{0.8}\text{Sb}_{0.2}$

*crystalline quality and trigonal orientation  
steadily improves with annealing*



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# Electrical Transport Characterization

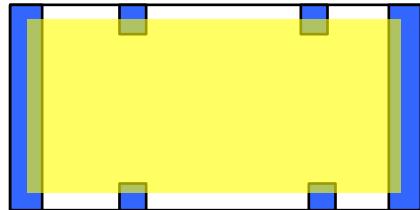
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- Since we utilize an insulating SiN cap, electrical contacts must be patterned **BEFORE** BiSb film deposition. Many metals fail because native oxides introduce an unacceptable contact resistance (e.g. Al, Ni, Ti).
- Many Bi-metal alloys (e.g. Au, Ag, Pd) have a eutectic point which causes melting well below our desired temperatures approaching 300 °C for optimal crystalline quality and trigonal orientation.

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- Many Bi-metal alloys (e.g. Au, Ag, Pd) have a eutectic point which causes melting well below our desired temperatures approaching 300 °C for optimal crystalline quality and trigonal orientation.
- Instead, we use 2 wt% Nd-doped Al. There is no eutectic point in the Bi-Al or Bi-Nd phase diagrams, and the Nd prevents surface oxidation yielding low resistance Ohmic contacts.

## Van der Pauw measurement



## Seebeck measurement

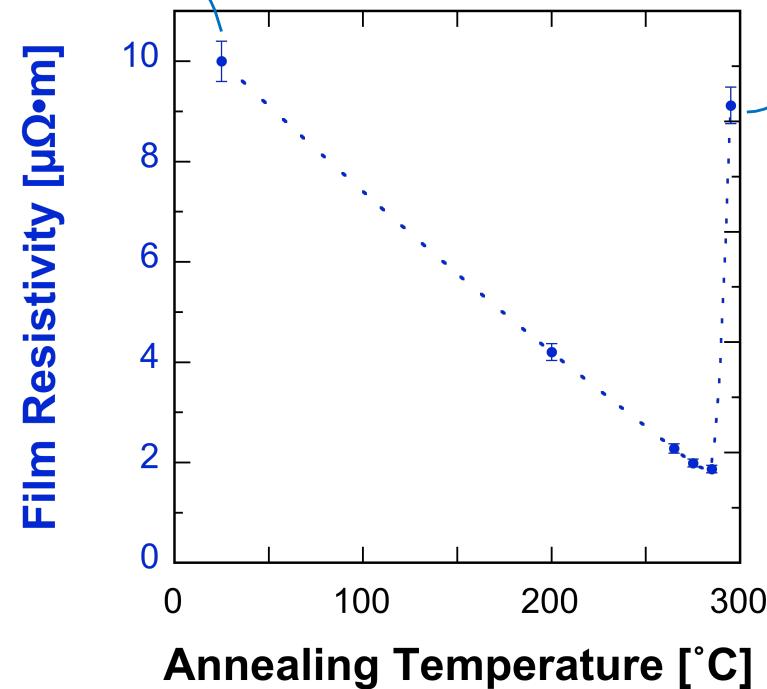
**Note:** all transport measurements will be made in the direction perpendicular to the trigonal orientation. This is NOT optimal, but still useful to correlate properties with microstructure.

# Thermoelectric Transport Properties

## Resistivity vs. Annealing Temperature

*all measurements at  $T = 300K$   
(perpendicular to the trigonal axis)*

*As-deposited  
films have the  
worst overall  
microstructure.*



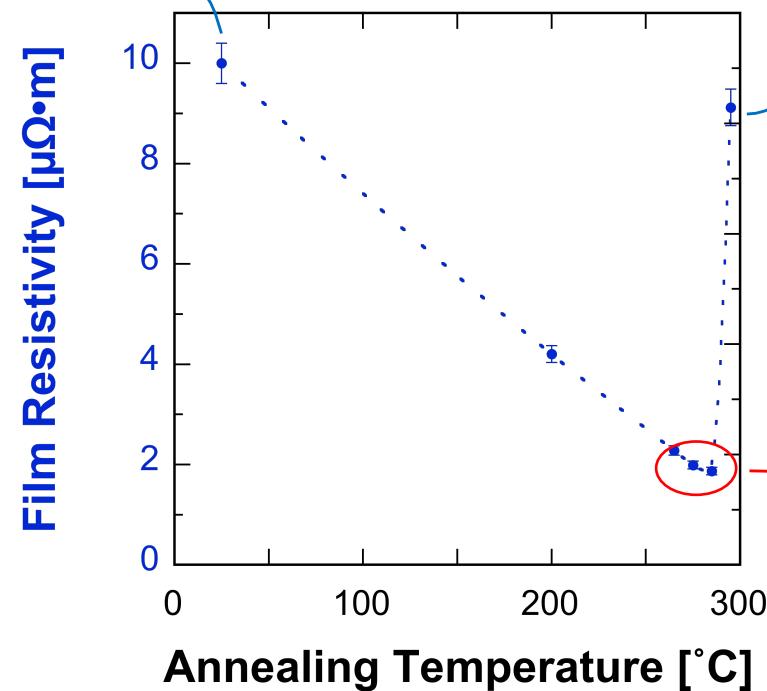
*Despite having the most  
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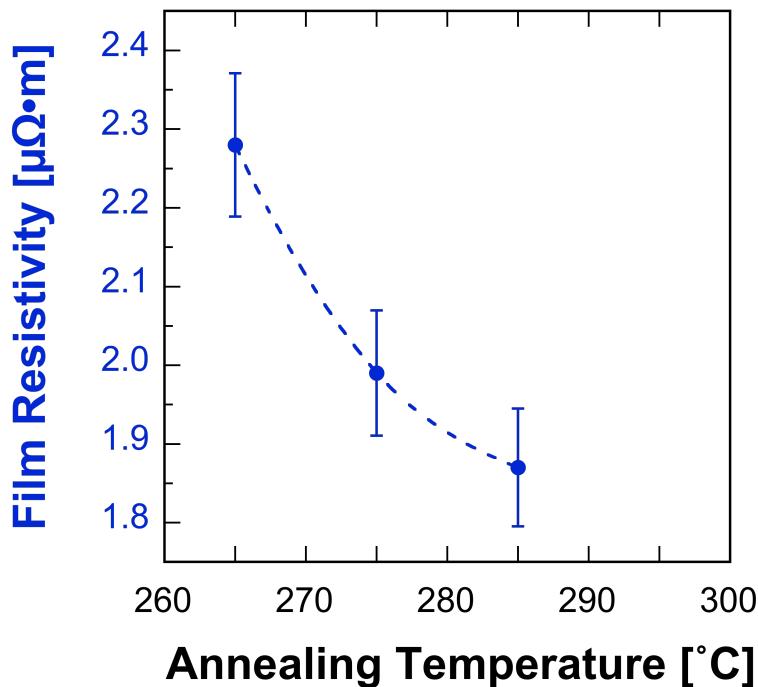
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*Focus on films  
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# Thermoelectric Transport Properties

## Resistivity vs. Annealing Temperature

*all measurements at  $T = 300K$   
(perpendicular to the trigonal axis)*



**Best  $\rho = 1.9 \mu\Omega\cdot m (\pm 4\%)$**

For reference:

**single crystal (18.2% Sb)**

**=  $1.6 \mu\Omega\cdot m$**  [Yim et al, SSE (1972) and Lenoir et al, JPCS (1996)]

**polycrystalline bulk disks (20% Sb)**

**=  $4.5 \mu\Omega\cdot m$**  [Malik et al, JAP (2012)]

**epitaxial MBE films (18.3% Sb, 1 μm thick)**

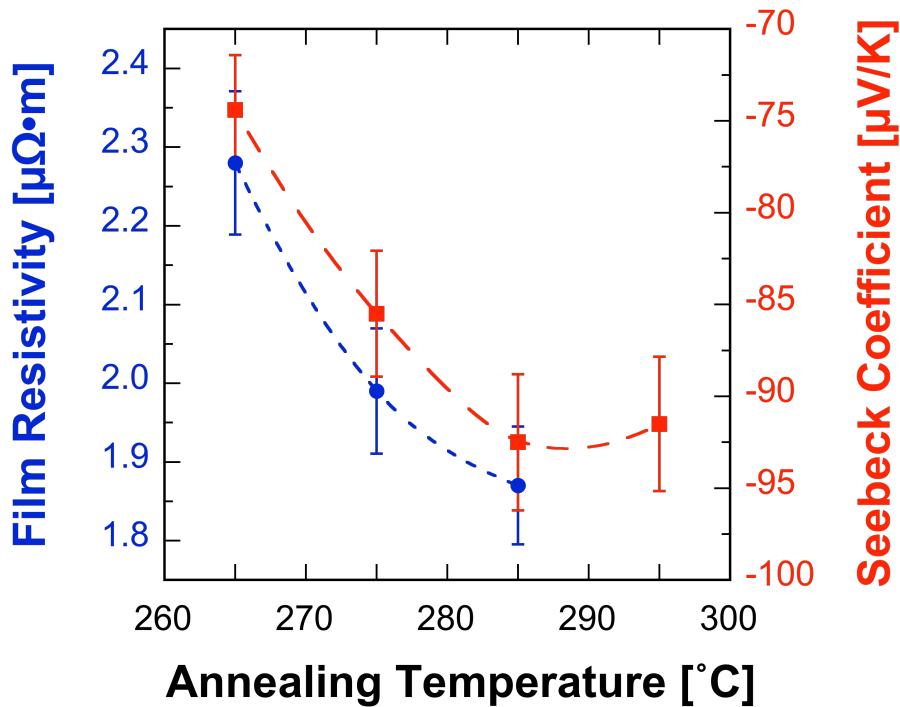
**=  $2.0 \mu\Omega\cdot m$**  [Cho et al. PRB (1999)]

# Thermoelectric Transport Properties

## Resistivity and Seebeck vs. Annealing Temperature

*all measurements at  $T = 300K$*

*(perpendicular to the trigonal axis)*



**Best  $S = -92.5 \mu\text{V/K}$  ( $\pm 4\%$ )**

For reference:

**single crystal (18.2% Sb)**

$= -76 \mu\text{V/K}$  [Yim et al, SSE (1972)]

**epitaxial MBE 1  $\mu\text{m}$  films (18.3% Sb)**

$= -55 \mu\text{V/K}$  [Cho et al. PRB (1999)]

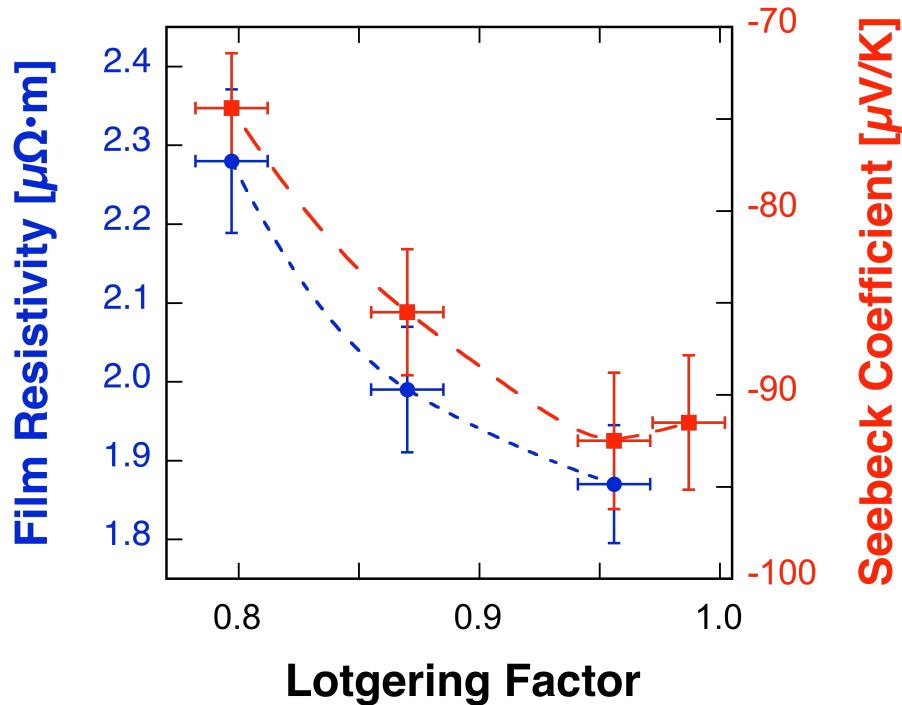
**100 nm evaporated films annealed to 150 °C (20% Sb)**

$= -97 \mu\text{V/K}$ , but have  $\rho = 20 \mu\Omega\cdot\text{m}$ ,  
*severely compromising performance*  
 [Malik et al, IEEE ICT (2003)]

# Correlating Transport with Microstructure

## Resistivity and Seebeck vs. Lotgering Factor

*all measurements at  $T = 300K$   
(perpendicular to the trigonal axis)*

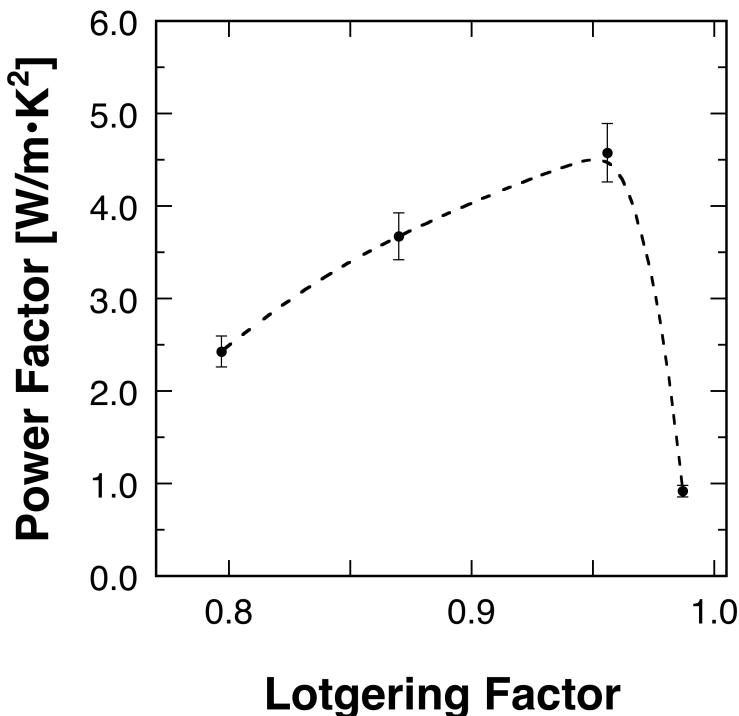


Both resistivity and Seebeck coefficient improve with trigonal orientation until annealing temperature gets too close to the alloy melting point.

# Correlating Transport with Microstructure

$$\text{Power Factor} = (\text{Seebeck})^2 / (\text{Resistivity})$$

*all measurements at  $T = 300K$   
(perpendicular to the trigonal axis)*



**Best PF = 4.6 mW/m·K<sup>2</sup> ( $\pm 7\%$ ).** This value is 28, 64, and 207% higher than the best reported single crystal, polycrystalline bulk alloy, and MBE film, respectively.

For reference:

**single crystal (18.2% Sb)**  
= 3.6 mW/m·K<sup>2</sup> [Yim et al, SSE (1972)]

**polycrystalline bulk disks (20% Sb)**  
= 2.8 mW/m·K<sup>2</sup> [Malik et al, JAP (2012)]

**epitaxial MBE films (18.3% Sb, 1  $\mu\text{m}$  thick)**  
= 1.5 mW/m·K<sup>2</sup> [Cho et al. PRB (1999)]

# Bi<sub>0.8</sub>Sb<sub>0.2</sub> Film Summary

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- Surface oxide presence depletes Sb from films during annealing, resulting in Bi-rich film compositions.
  - SiN cap layer prevents Sb-oxide formation.
  - enables correlation of microstructure and transport properties.
- If electrical contacts are deposited prior to thermal annealing, then
  - helpful if they neither oxidize nor react with Bi<sub>1-x</sub>Sb<sub>x</sub>, such as Nd-doped Al.
- Microstructural quality improves with annealing temperature.
  - crystalline quality (XRD-FWHM)
  - trigonal orientation (LF)
  - morphology (SEM), until near the alloy melting point.
- Film resistivity and thermopower both systematically improve with overall microstructural quality.
  - results in TE power factors better than those previously reported for thin films, polycrystalline bulk alloys, and even single crystals.