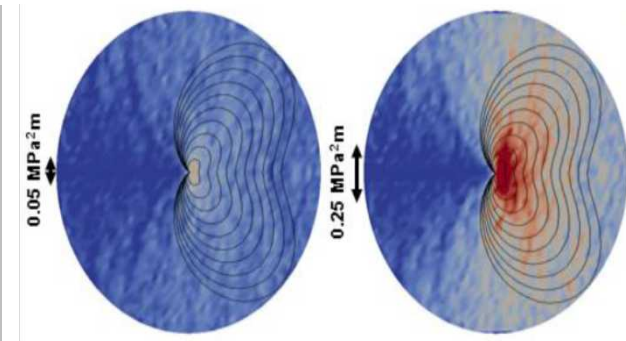
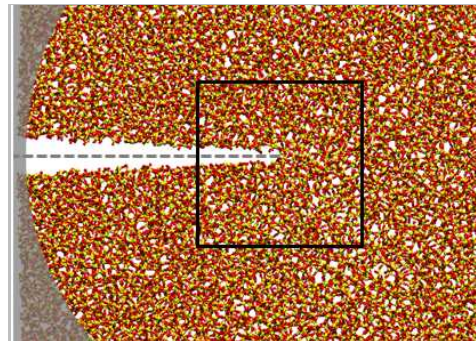
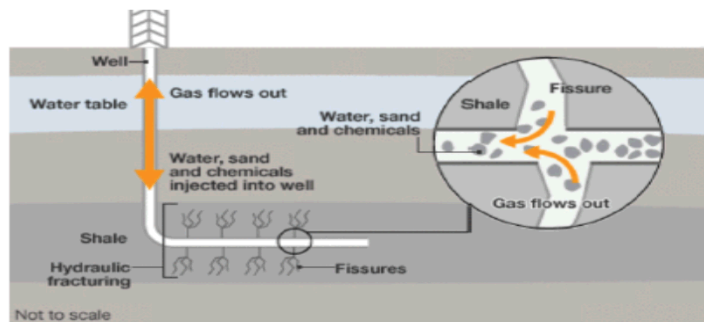


Hydraulic Fracturing in Shale Gas Reservoirs

Shale Gas Reservoirs



# Nanoscale Stress-Corrosion of Silicate Glass in Aqueous Solutions

Louise J. Criscenti, Jessica M. Rimsza,  
Edward N. Matteo, and Reese E. Jones

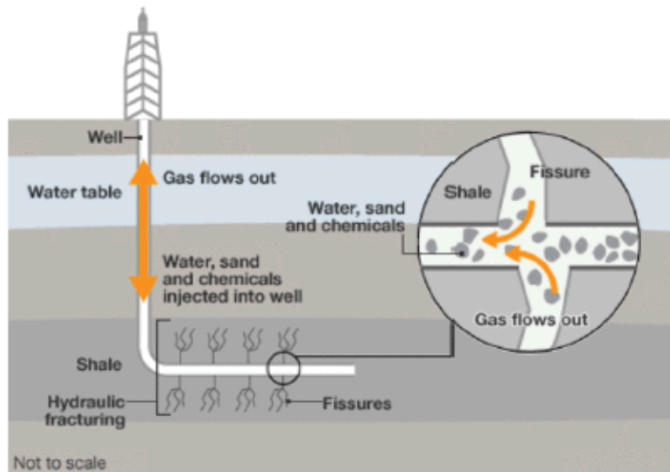


Sandia National Laboratories is a multimission laboratory managed and operated by National Technology and Engineering Solutions of Sandia, LLC, a wholly owned subsidiary of Honeywell International, Inc., for the U.S. Department of Energy's National Nuclear Security Administration under contract DE-NA0003525.

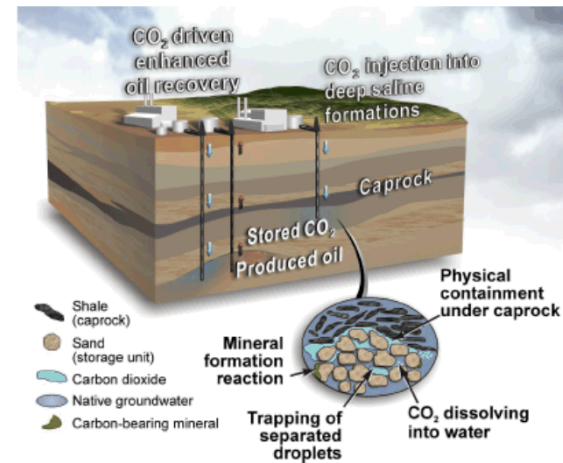
# Motivation

## Hydrofracking

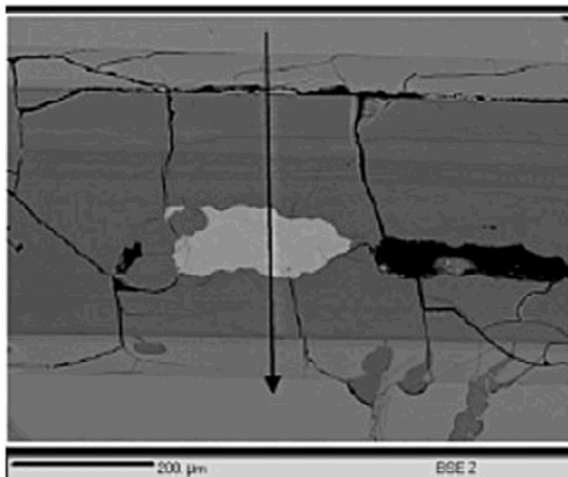
Shale gas extraction



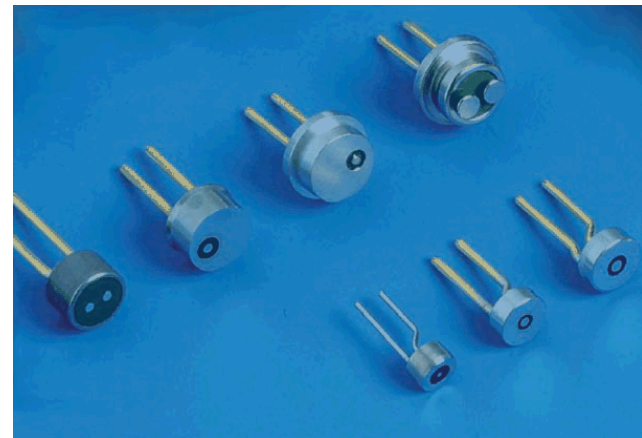
## Sequestration



Crack in glass:  
analog for glass waste form



Airbag feedthrough igniters

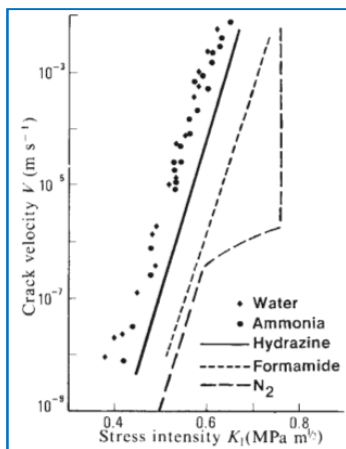


- To develop a fundamental atomistic-level understanding of the chemical-mechanical mechanisms that control subcritical cracks in low-permeability geomaterials.
- To link atomic-scale insight to macroscale observables and directly address how chemical environment affects mechanical behavior.
- **Why Atomistic Simulation?**
  - Cracks start at the atomistic scale – a crack tip will be initiated by the breaking of bonds (e.g., Si-O) at the rock-fluid interface.
  - Crack tip formation and crack propagation is impacted by interfacial fluid and surface chemistry (e.g., development of surface charge, impact of adsorbed species along fracture surface).

# Chemical Effects on Subcritical Rock Fracture

## Stress-Corrosion Cracking

### Silica Glass

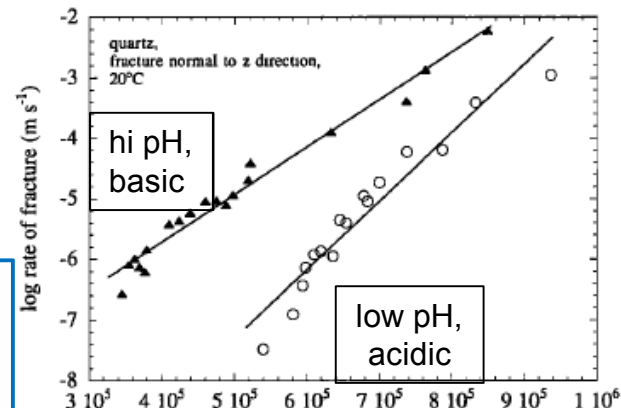


T.A. Michalske & S.W. Freiman, *Nature* (1982)

$K_I$  = measure of  
stress field at crack  
tip.

Orders of magnitude  
Difference in the rate  
of fracture with  
chemical environment

### Quartz



$K_I$  = stress intensity factor ( $\text{N m}^{-3/2}$ )

P. M. Dove, *J. Geophys. Res.* (1995)

Dove (1995) equation for rate of Si-O bond rupture

$$r_{\text{Si-O}} = \underbrace{A_{\text{H}_2\text{O}}^* \left( \exp \frac{-\Delta H_{\text{H}_2\text{O}} + b_{\text{H}_2\text{O}} K_I}{RT} \right) (\theta_{\text{Si-O}}^{\text{H}_2\text{O}})}_{>\text{Si-O-Si}< + \text{H}_2\text{O} = * = 2>\text{SiOH}} + \underbrace{A_{\text{OH}^-}^* \left( \exp \frac{-\Delta H_{\text{OH}^-} + b_{\text{OH}^-} K_I}{RT} \right) (\theta_{\text{Si-O}}^{\text{OH}^-})}_{>\text{Si-O-Si}< + \text{OH}^- = * = >\text{SiOH} + >\text{SiO}^-}$$

Without  $bK_I$  term, this expression would strictly describe quartz **dissolution**

# Objectives

- To calculate fracture toughness of dry silica glass from atomistic molecular simulation data
  - Start with a simple, well-studied isotropic material
  - Upscale atomistic data to macroscopic observable (fracture toughness)
- To calculate fracture toughness of wet silica glass from atomistic molecular simulation data
- To investigate the impact of different simple aqueous solution compositions on fracture propagation and toughness:
  - Cations:  $\text{Li}^+$ ,  $\text{Na}^+$ ,  $\text{Ca}^{2+}$
  - Anions:  $\text{OH}^-$ ,  $\text{Cl}^-$
- To compare modeling and experimental results.

## Griffith Criterion:

$$\sigma_{cr} = \left( \frac{2E\gamma}{\pi c} \right)^{\frac{1}{2}}$$

**Strain energy release rate** ( $G$ ) describes the energy stored in material prior to fracture

A linear elastic solution for a slit crack in plane strain mode I leads to:

$$G_{IC} = J_{IC} = \frac{K_{IC}^2 (1 - \nu)^2}{E} = 2\gamma_s$$

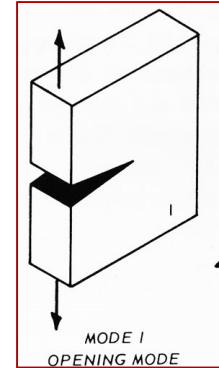
(brittle material)

$$G_{IC} = G_{diss} + 2\gamma_s$$

(material with inelastic behavior)

**Fracture toughness ( $K_{IC}$ ):** the energy required to propagate a crack in a material.

$\sigma_{cr}$  = critical stress  
 $E$  = Young's modulus  
 $c$  = crack length  
 $\gamma$  = surface energy  
 $K_I$  = stress intensity factor  
 $\nu$  = Poisson's ratio  
 $G$  = strain energy release rate



**J-integral:** a path independent contour integral used to calculate  $G$  for monotonic loading

$$J = \int_{\partial\Omega} S \cdot dA$$

Eshelby stress field:

$$S = WI - H^TP$$

$I$  is identity tensor

$H$  = displacement gradient, measure of strain

$P$  = stress

$W$  = energy

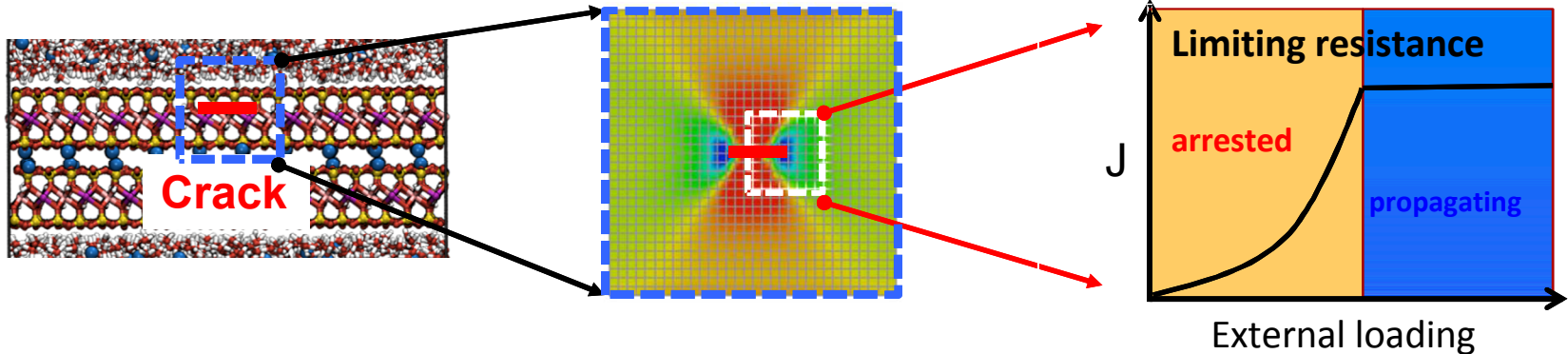
# Mechanical Fracture Models

## Upscaling Metrics: Atomistic to Continuum (AtC)

Molecular simulations sample atomic displacements, energies & forces

Eshelby stress fields use atomic data to characterize local energy available to move the crack (*in red*)

The J-integral is formed from the divergence of the Eshelby stress and determines if a crack will propagate

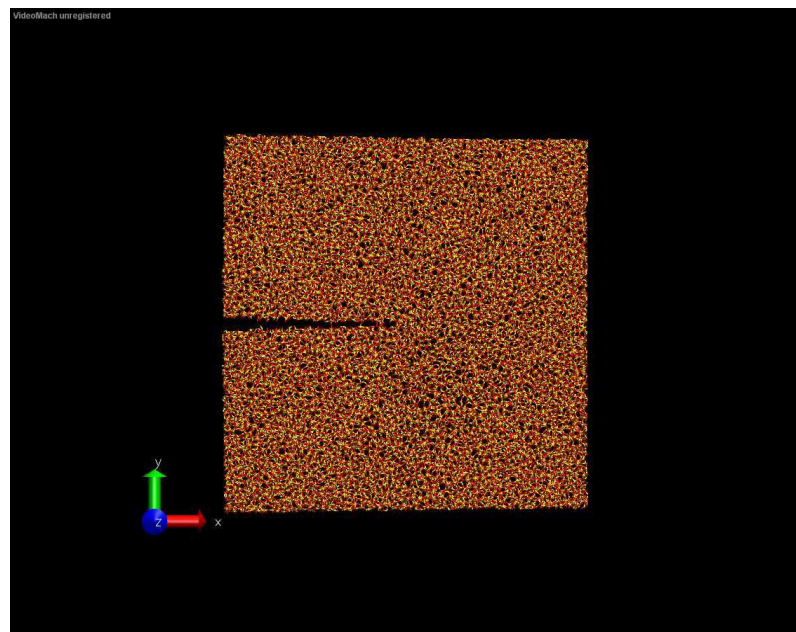
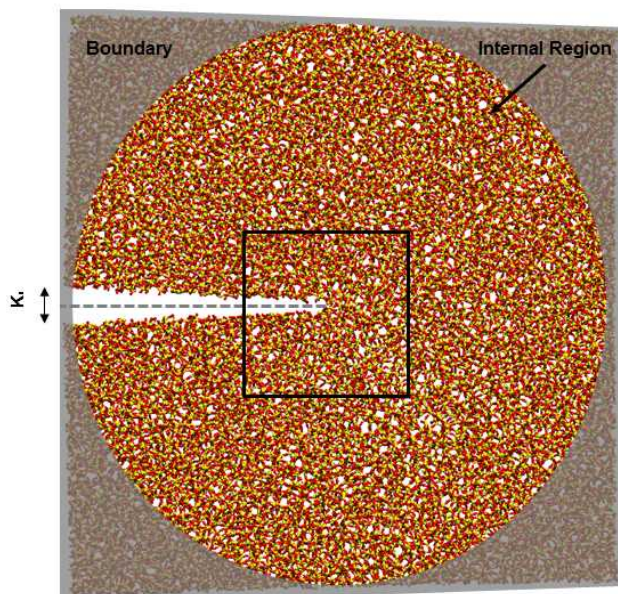


**Jones & Zimmerman: First to successfully apply continuum fracture theory to atomistic systems to estimate fracture toughness**

- The J-integral provides the characteristic crack tip driving force regardless of system configuration.
- **When J reaches the material's fracture toughness ( $J_c$ ), crack propagation begins.**
- $J/J_c$  can be measured experimentally using geometries and procedures specified in standards (e.g. ASTM E1820 - Standard Test Method for Measurement of Fracture Toughness)

# Simulation Set-Up for Dry Silica System

- ReaxFF Force Field, LAMMPS MD code (Plimpton, XXXX)
- Amorphous silica, system size: 38,400 atoms (143 Å x 143 Å x 28 Å); Melt and quenched from  $\beta$ -cristobalite system; 12 replicas
- A slit crack is formed creating a singularity with highly concentrated stress
- Boundary atoms are fixed; Far-field loading as Mode I fracture
- Interior atoms are allowed to move freely
- Stress is introduced iteratively by increasing the crack width
- Atomic positions adjust to accommodate the added stress



# Silica Fracture and Energy Dissipation

- Fracture propagates in distinct steps when the stress at the crack tip exceeds the strength of the material
- Perfectly brittle fracture will have no dissipation energy, with all energy used to propagate the fracture
- Dissipation energy calculated from change in energy of the system over the added surface area

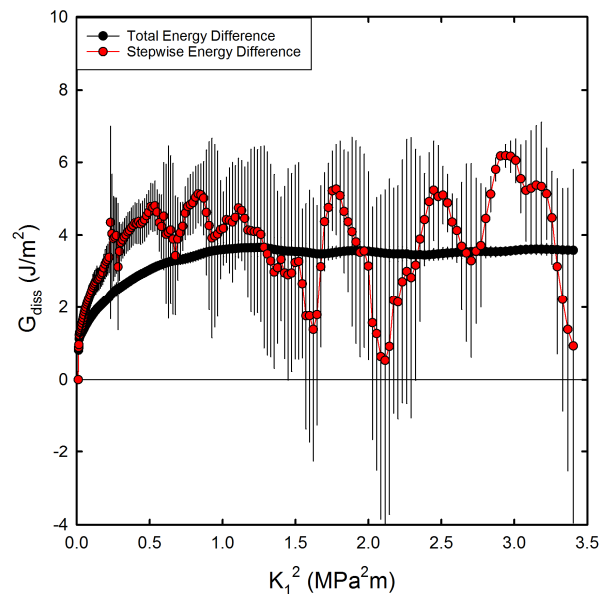
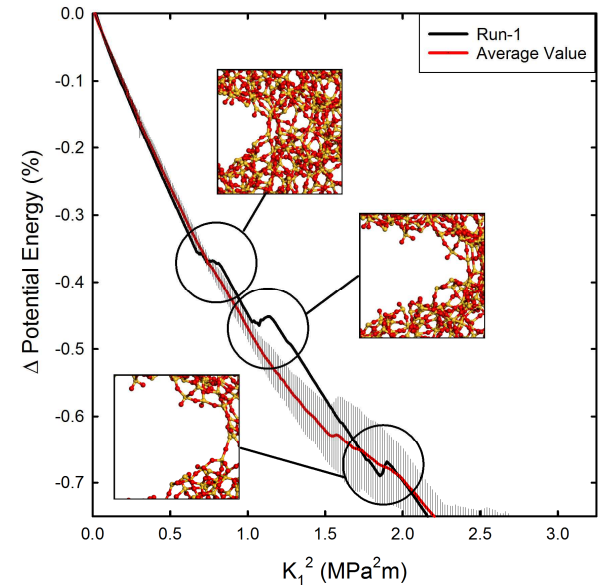
$$\frac{\Delta U}{\Delta S_A} = G_{diss}$$

- Local inelastic behavior as energy is introduced into the silica during loading and not completely dissipated once fracture occurs

$$J_{IC} = G_{IC} = G_{diss} + 2\gamma_s$$

- Surface energies ( $\gamma_s$ ): 1.2-2 J/m<sup>2</sup>

Rimsza, J. M. et al. *Langmuir* (2017).

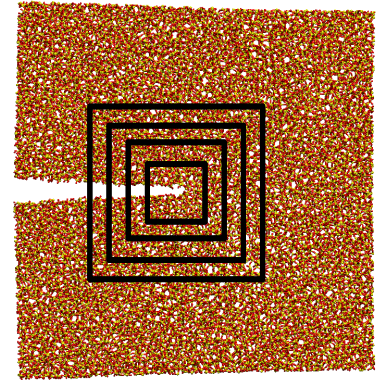


$$G_{diss} = 3.5 \text{ J/m}^2$$

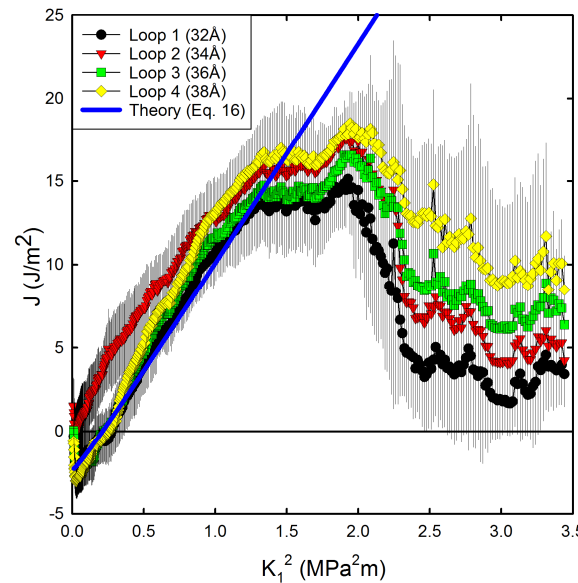
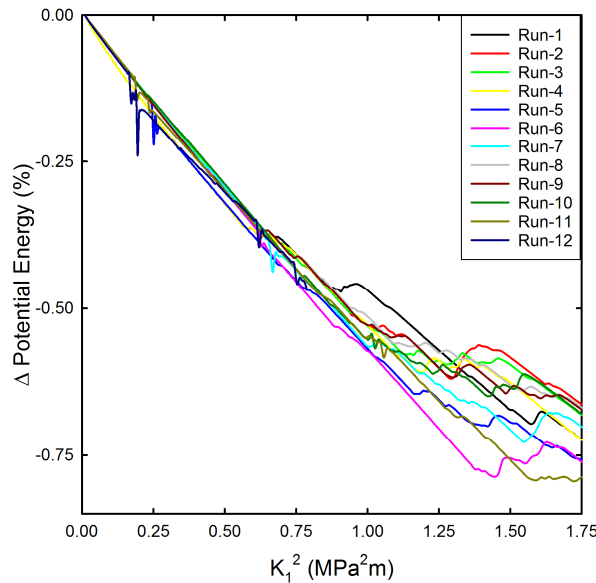
Estimated  $J_{IC}$  values  
are ~5.9-7.5 J/m<sup>2</sup>

# J-Integral Calculation for Dry Silica

- Calculated via the AtC method through coarse graining energy, displacement, and stress  
*Jones, Reese E., et al. J. Phys.: Condens. Matter (2010)*
- J-integral converges at loop sizes of  $\sim 3$  nm approximates the size of the inelastic zone



Schematic of increasing loop sizes for J-integral convergence test

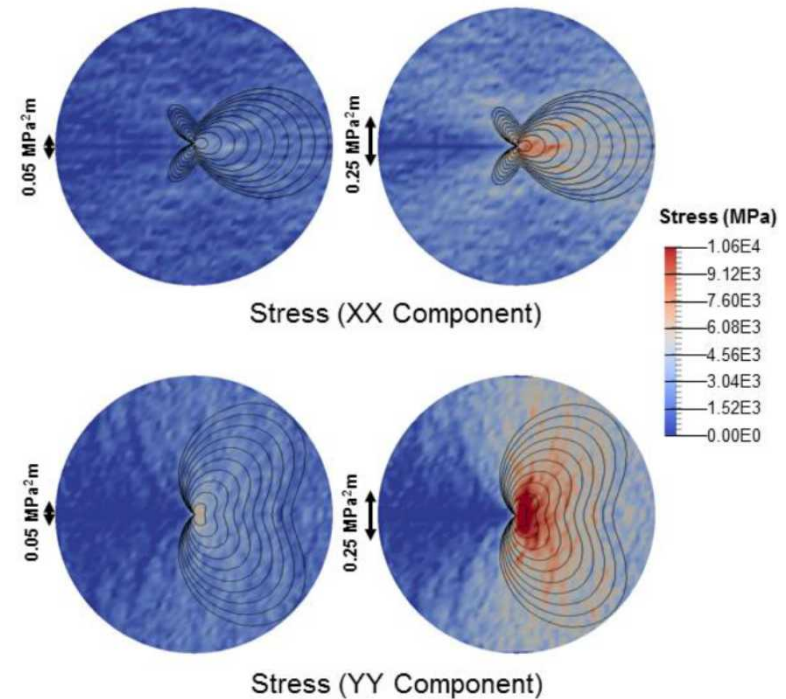
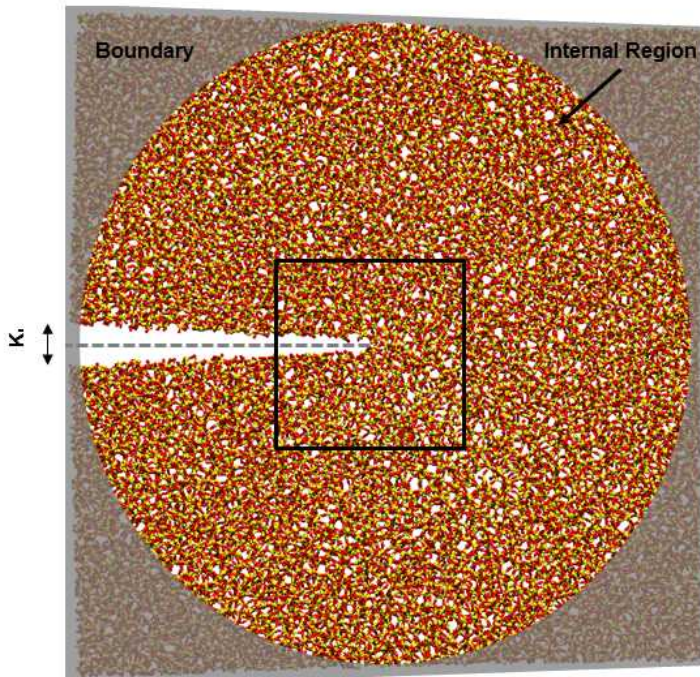


J-integrals with converged loop sizes

Calculated  
 $K_{IC} = 0.76 \pm 0.16 \text{ MPa}\sqrt{\text{m}}$   
 $J_{IC} = 6.16 \pm 4.34 \text{ J/m}^2$

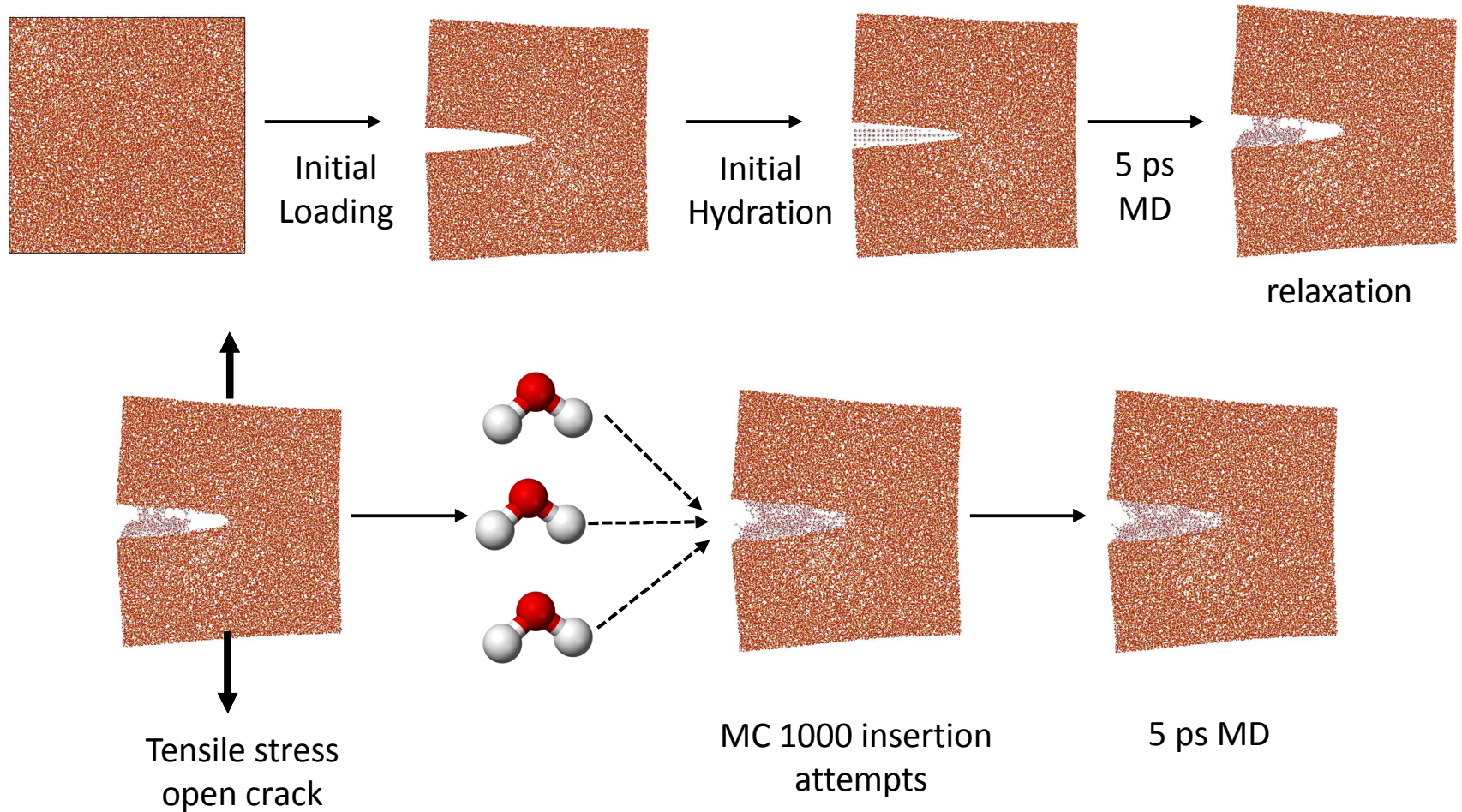
Experimental  
 $K_{IC} = 0.78 \pm 0.04 \text{ MPa}\sqrt{\text{m}}$

# Simulated Stress Fields for Dry Silica



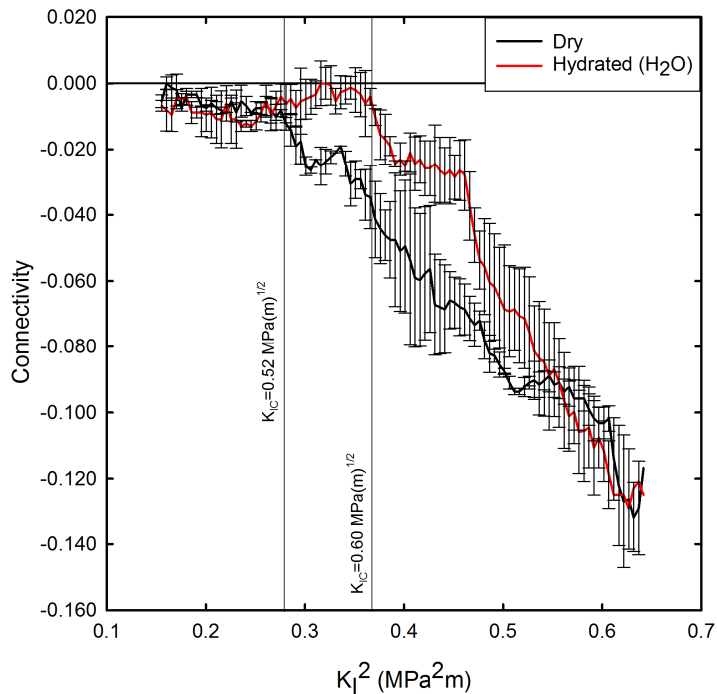
Crack tip stress fields in amorphous silica  
at two loading levels ( $K_I^2 \sim 0.05$  and  $0.25 \text{ MPa}^2\text{m}$ ).  
Black contours are from linear elastic solution.

# Silica Fracture in Water: Methodology

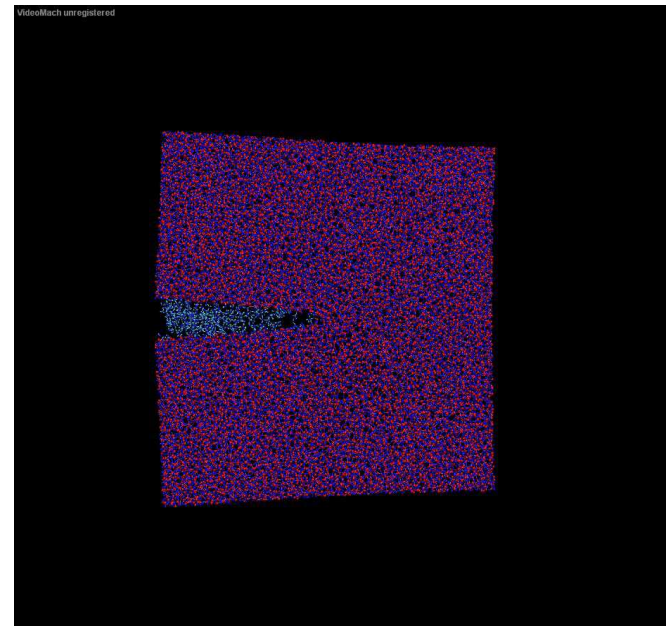


# ReaxFF: Fracture of Hydrated Silica

- Water placed inside the crack followed by far field loading
- Dry silica fractures with less loading ( $\sim 0.52 \text{ MPa}\sqrt{\text{m}}$ ) compared with wet systems ( $\sim 0.60 \text{ MPa}\sqrt{\text{m}}$ )

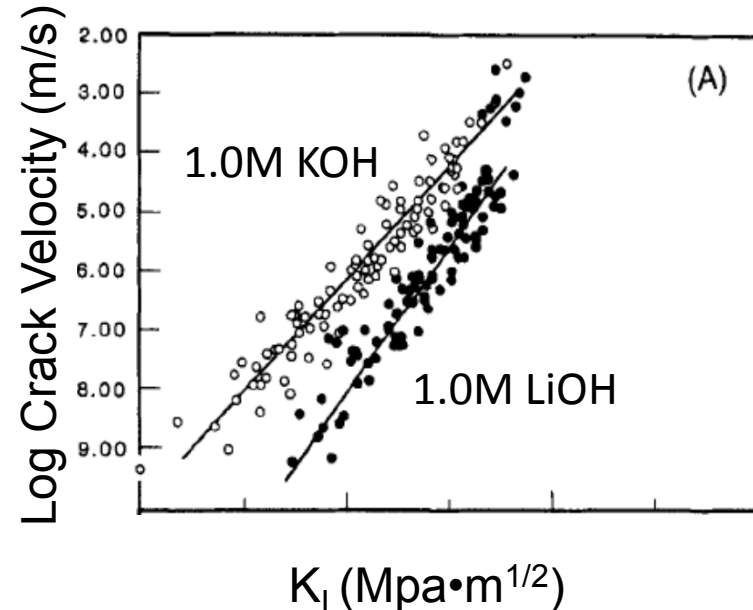
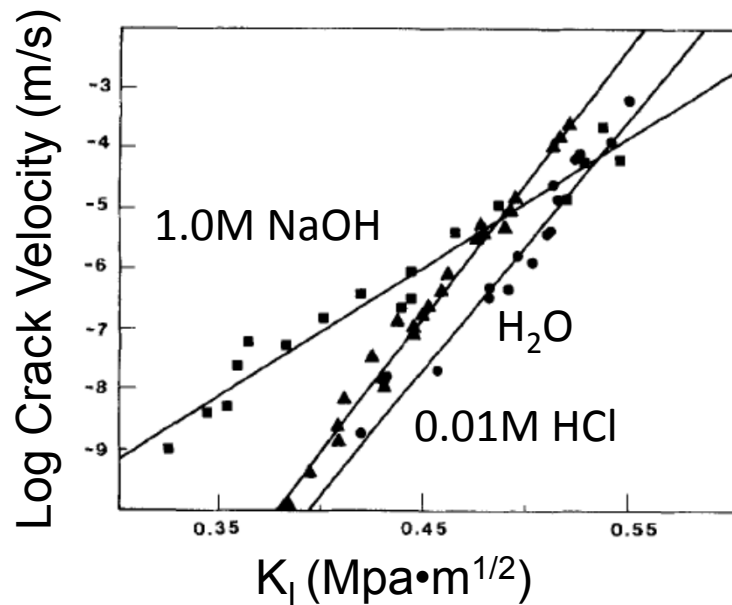


Changing connectivity with loading to identify when a fracture propagation event occurs in dry and hydrated systems



Fracture of hydrated silica  
Colors: water (blue), silica (pink and purple)

# Effect of Counterions on Crack Growth in Vitreous Silica



- The chemical activity of hydroxide is affected by the nature of the cation due to ion association.
- Hydroxide can penetrate the solvation shell of molecules.
- LiOH association decreases OH<sup>-</sup> activity in solution more than NaOH association.
- Dependence of applied stress on cation identity: NaOH ~ KOH ~ CsOH > LiOH

0.1M NaOH    pK = -0.7  
0.1M LiOH    pK = -0.08

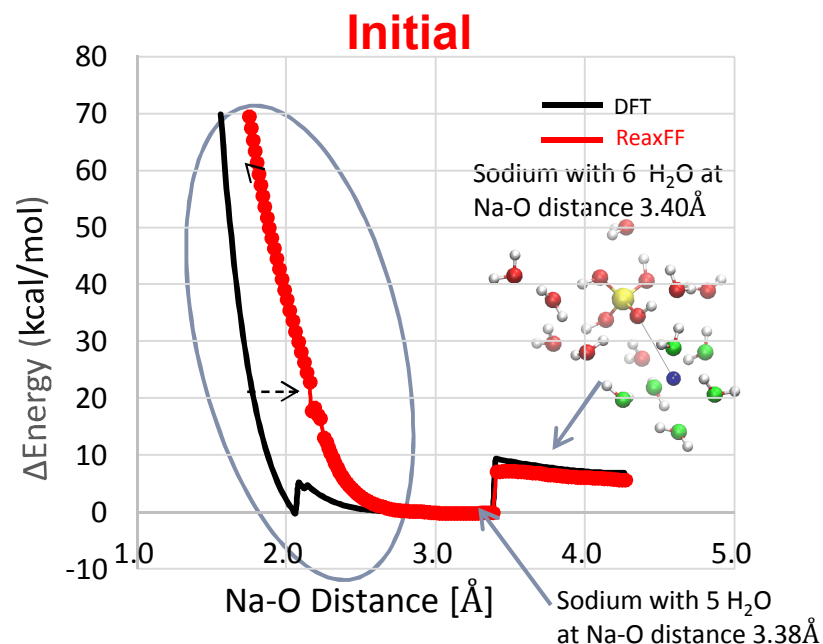
1M NaOH    9% Associated  
1 M LiOH    26% Associated

*White et al., 1987*

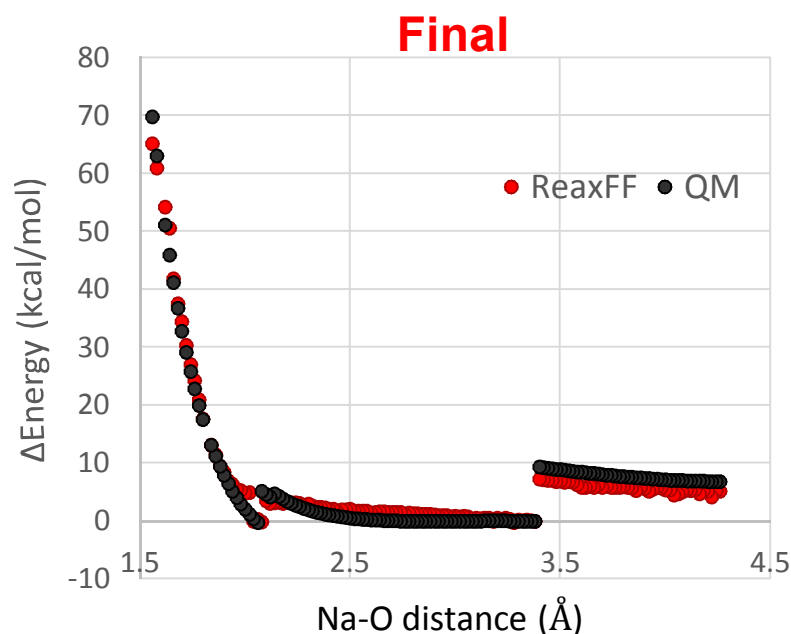
# REAXFF Development

Adri van Duin, Penn State

- Parameters for aqueous electrolytes under development at PennState
- NaOH, KOH, LiOH: Exhibits differences in association constants that are observed experimentally
- Collaboration between PennState and Sandia to develop parameters for  $\text{Na}^+\text{-H}_2\text{O-SiO}_2$ ,  $\text{Li}^+\text{-H}_2\text{O-SiO}_2$

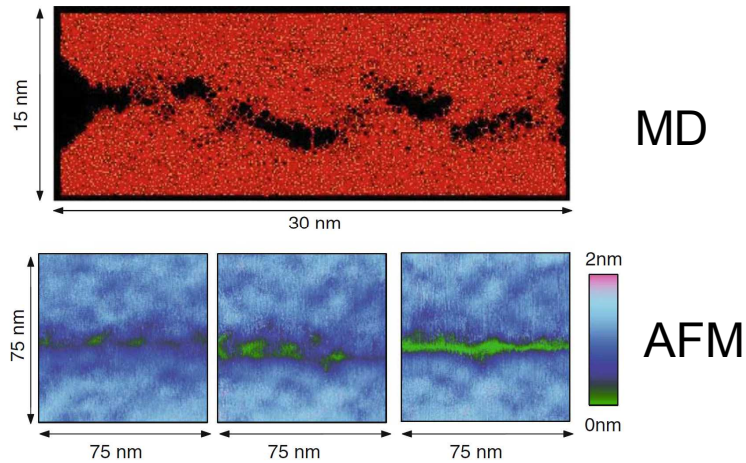


- Good agreement in 2.5-4.5 Å range
- ReaxFF is too repulsive below 2.5 Å

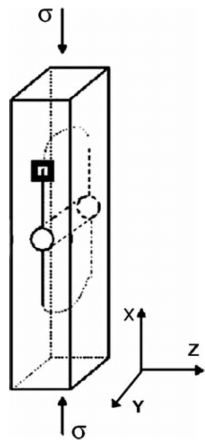


- Excellent agreement through the entire range, including energy jumps around 3.38 Å and 2.1 Å associated with H-bond configuration changes

# Experimental Validation of $K_{IC}$

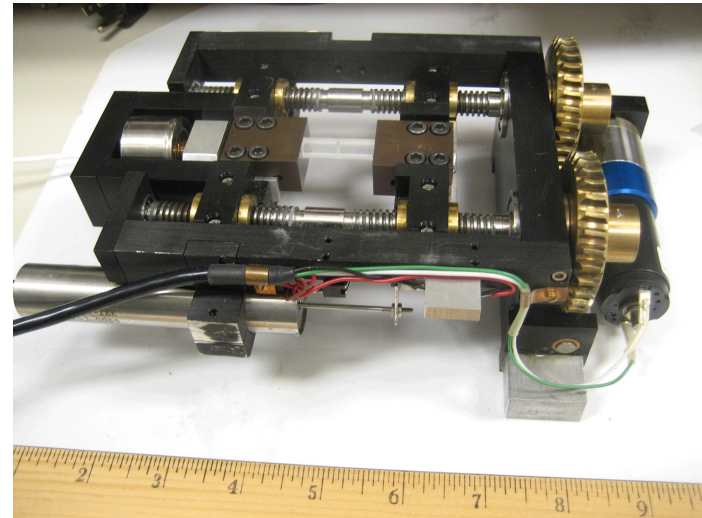


*Bonomay et al. Int. J. Fract. 2006*



DCDC = Double Cleavage  
Drilled Compression

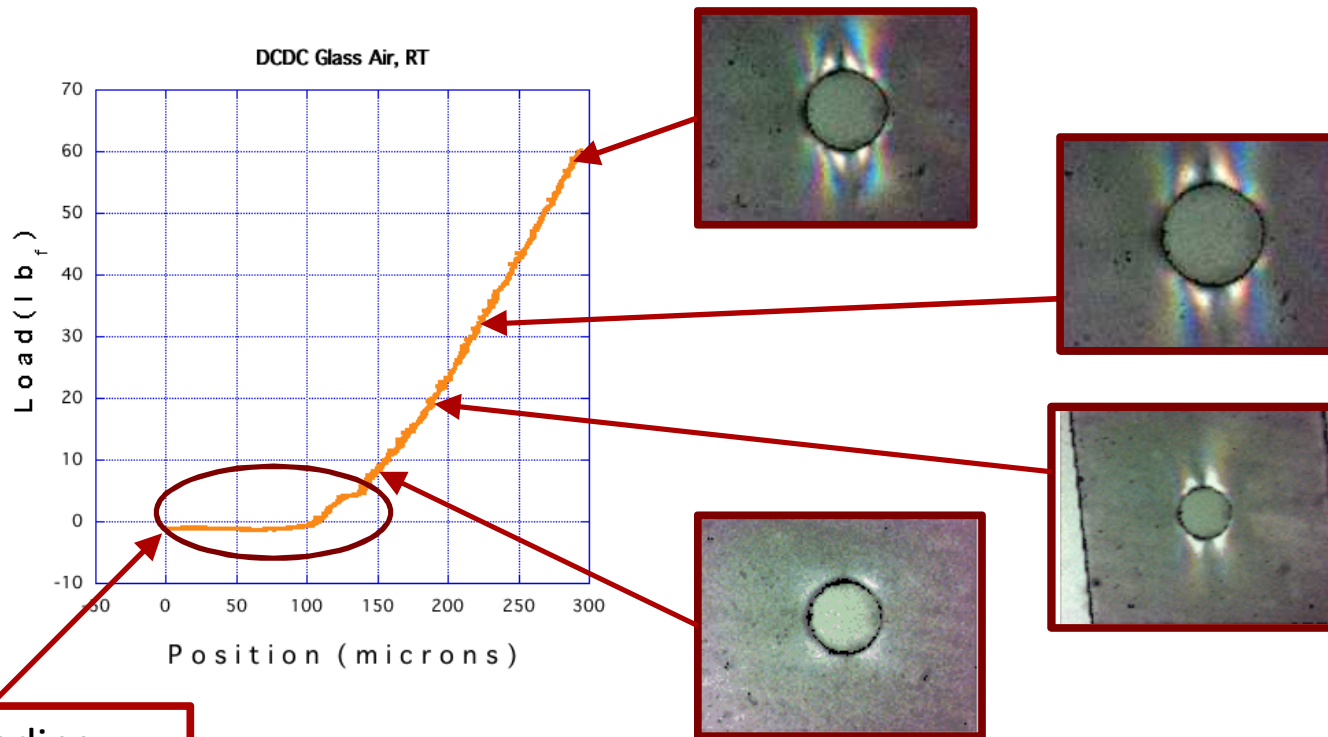
*Technique may also be  
implemented with  
confocal microscope.*



Load Frame

*Review articles: Ciccotti, J. Phys. D 2009, Ciccotti et al. JNCS 2008*

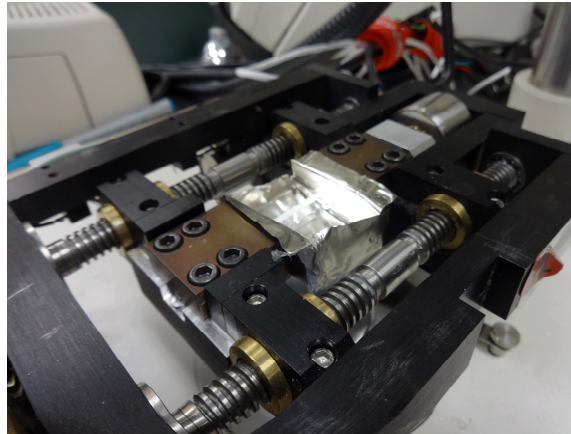
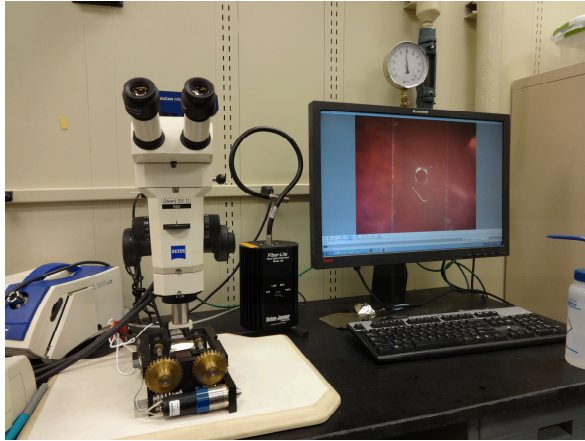
# Imaging Capabilities and Preliminary Results for Dry Silica



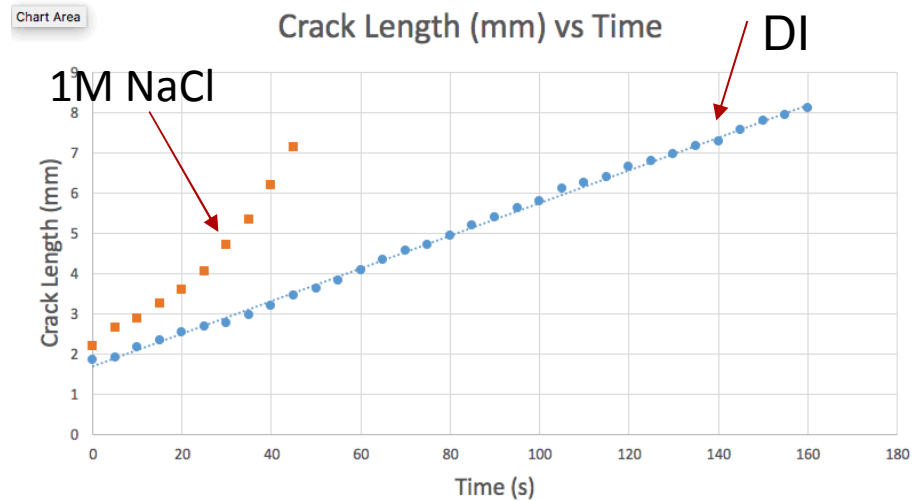
Sample loading  
and compliance of  
nickel foil

- Optical profilometry for post-test analysis and preliminary imaging of load frame tests of DCDC samples
- Polarized light microscopy for visualizing stress fields; will use Moiré analytical methods for stress field quantification

# Experimental Results in Solution



- Tested design for loading in liquids
- Repeatable loading of DCDC glass samples in DI water
- Air, DI, and 0.5 NaCl tested
- Upgraded test equipment for load frame and imaging
  - Software, updated laptop (Windows 10)
  - Time lapse software
  - Upgraded USB microscope

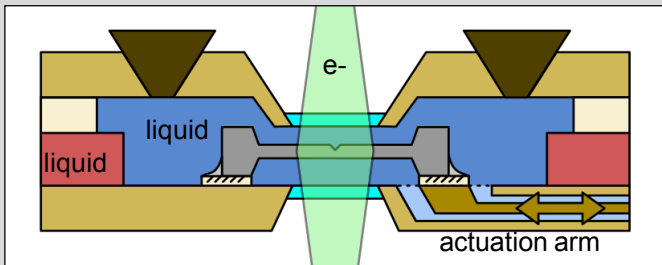


# Different SNL Project

## Developing Liquid-Mechanical TEM Platform

- First-of-its-kind MEMS device for quantitative testing the mechanical properties of materials within a controlled (chemical and temperature) liquid environment that is coupled with high-resolution real-time imaging
- Through BES support at CINT, this capability will be available to the international research community

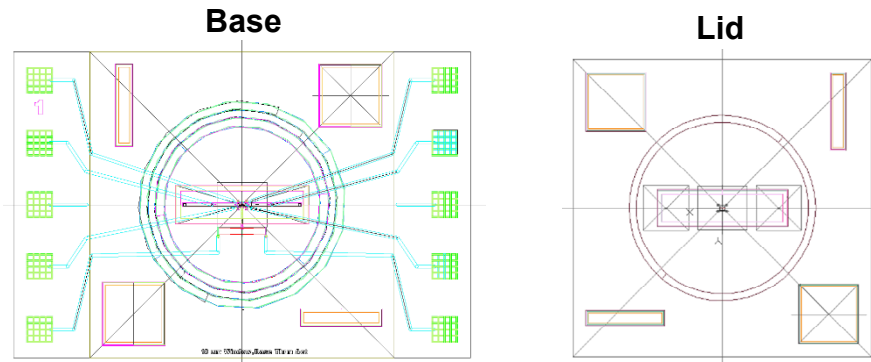
### Generation 1 Concept



- Buried movable arm attached to sample
- Sample suspended over large pads
- Heating of liquid seal-ring
- Prevention of sample/window stiction
- SAM in movable arm and liquid seal-ring
- Low and high force actuators

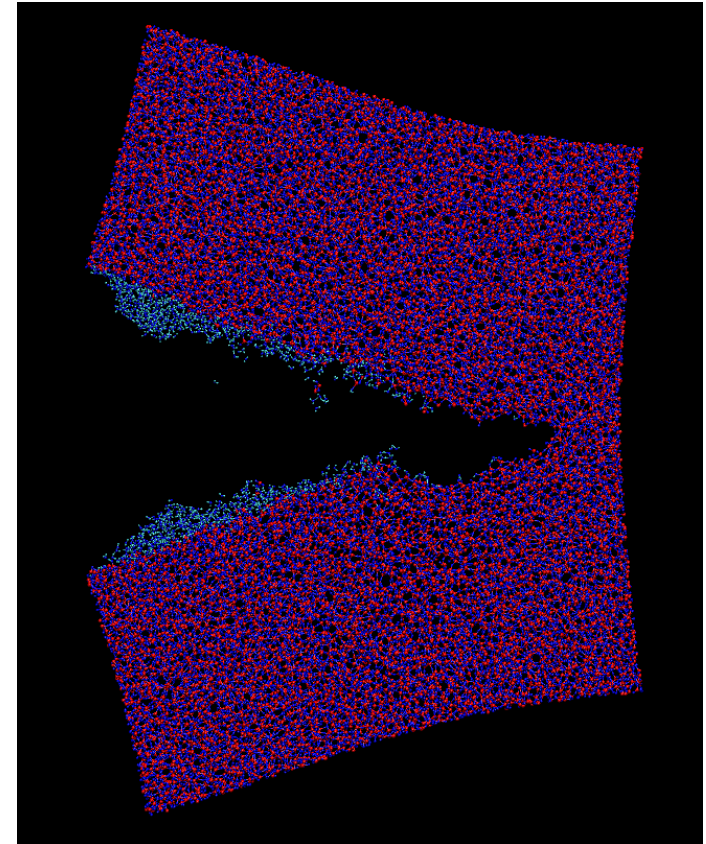
### *Liquid-Mechanical MEMS Device*

- Nano Fracking TEM Base: 15 modules, 11 different designs, 240 usable die per wafer
  - 11 designs for testing of heaters, window dimensions, window thickness, sample placement, actuation strength, single vs. dual actuation, and capacitive sensing
- Nano Fracking TEM Lid: 25 modules, 4 different designs, 400 usable die per wafer
  - 4 different designs based on window dimensions and thickness



# Summary and Conclusions

- First time atomistic simulation data has been used to calculate the macroscopic J-integral and crack stress fields from atomistic data for an oxide.
- First time ReaxFF has been used to investigate subcritical crack propagation using a slit crack and Mode I tension.
- First time that a molecular-scale simulation method has been developed for stress-corrosion cracking in DI water.
- First time that DCDC experiments have been conducted in environmental chambers – new experimental design
- In simulations for dry silica, a localized high stress inelastic region develops prior to fracture with an estimated radius between 3-3.2 nm.
- The calculated  $K_{IC}$  of  $\sim 0.76 \text{ MPa}\sqrt{\text{m}}$  is consistent with experimental data reported in the literature.
- Preliminary data is consistent with conceptual models and previous experimental data.



Snapshot of fracture silica in a hydrated environment

## Acknowledgements

*This work was fully supported by the Laboratory Directed Research and Development (LDRD) program of Sandia National Laboratories.*