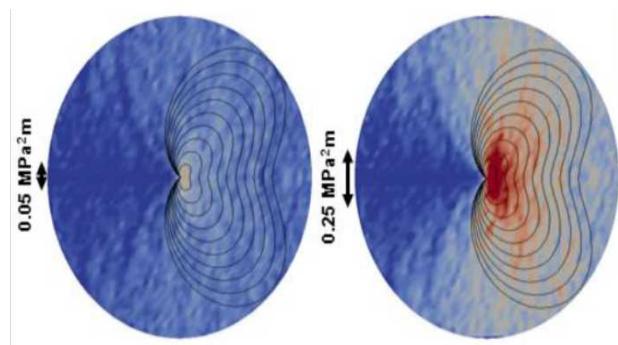
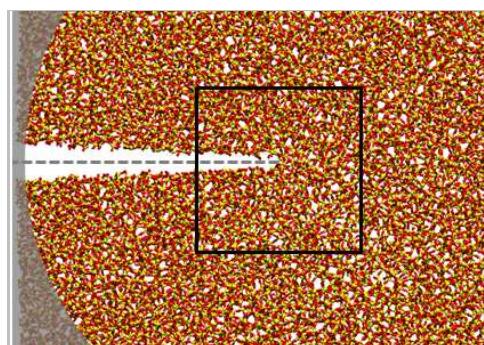
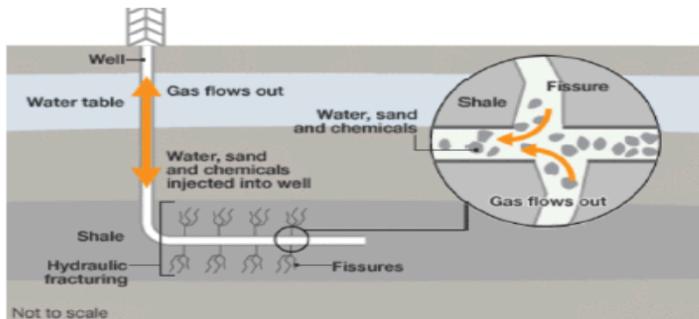


Hydrofracturing and gas flow in shale  
and sandstone reservoirs

Hydrofracturing  
and gas flow in shale  
and sandstone reservoirs



# Nanoscale Stress-Corrosion of Silicate Glass in Aqueous Solutions

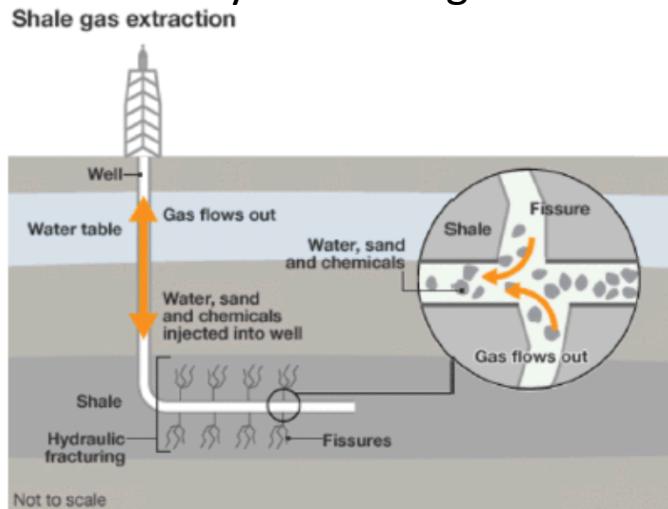
Louise J. Criscenti, Jessica M. Rimsza,  
Edward N. Matteo, and Reese E. Jones



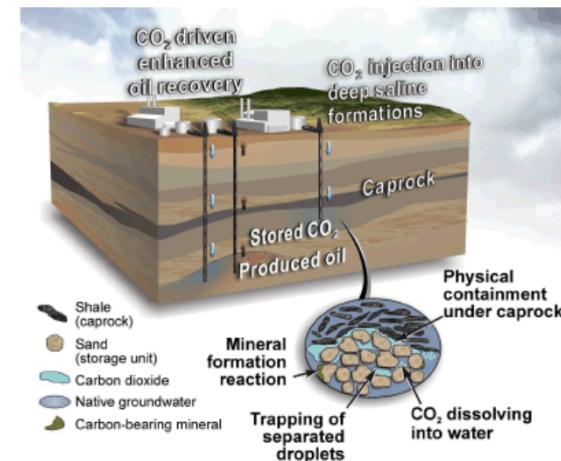
Sandia National Laboratories is a multimission laboratory managed and operated by National Technology and Engineering Solutions of Sandia, LLC, a wholly owned subsidiary of Honeywell International, Inc., for the U.S. Department of Energy's National Nuclear Security Administration under contract DE-NA0003525.

# Motivation

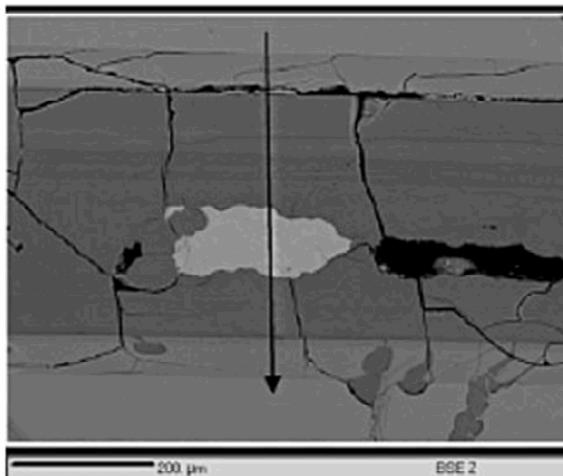
## Hydrofracking



## Sequestration



Crack in glass:  
analog for glass waste form



Airbag feedthrough igniters



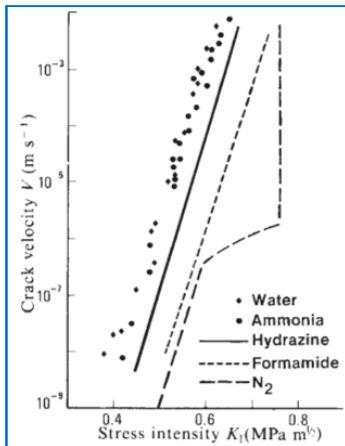
# Overview

- To develop a fundamental atomistic-level understanding of the chemical-mechanical mechanisms that control subcritical cracks in low-permeability geomaterials.
- To link atomic-scale insight to macroscale observables and directly address how chemical environment affects mechanical behavior.
- **Why Atomistic Simulation?**
  - Cracks start at the atomistic scale – a crack tip will be initiated by the breaking of bonds (e.g., Si-O) at the rock-fluid interface.
  - Crack tip formation and crack propagation is impacted by interfacial fluid and surface chemistry (e.g., development of surface charge, impact of adsorbed species along fracture surface).

# Chemical Effects on Subcritical Rock Fracture

## Stress-Corrosion Cracking

### Silica Glass

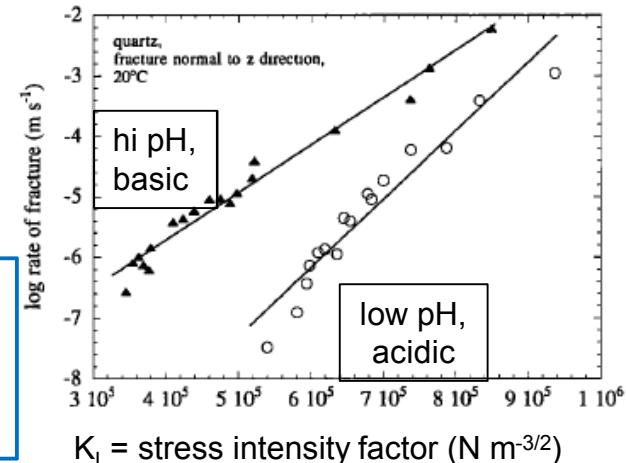


$K_I$  = measure of  
stress field at crack  
tip.

Orders of magnitude  
Difference in the rate  
of fracture with  
chemical environment

T.A. Michalske & S.W. Freiman, *Nature* (1982)

### Quartz



$K_I$  = stress intensity factor ( $\text{N m}^{-3/2}$ )

P. M. Dove, *J. Geophys. Res.* (1995)

Dove (1995) equation for rate of Si-O bond rupture

$$r_{\text{Si}-\text{O}} = A_{\text{H}_2\text{O}}^* \left( \exp \frac{-\Delta H_{\text{H}_2\text{O}} + \mathbf{b}_{\text{H}_2\text{O}} K_I}{RT} \right) (\theta_{\text{Si}-\text{O}}^{\text{H}_2\text{O}}) + A_{\text{OH}^-}^* \left( \exp \frac{-\Delta H_{\text{OH}^-} + \mathbf{b}_{\text{OH}^-} K_I}{RT} \right) (\theta_{\text{Si}-\text{O}}^{\text{OH}^-})$$

$\text{>Si-O-Si<} + \text{H}_2\text{O} = * = 2\text{>SiOH}$        $\text{>Si-O-Si<} + \text{OH}^- = * = \text{>SiOH} + \text{>SiO}^-$

Without  $\mathbf{b}K_I$  term, this expression would strictly describe quartz *dissolution*

# Objectives

- To calculate fracture toughness of dry silica glass from atomistic molecular simulation data
  - Start with a simple, well-studied isotropic material
  - Upscale atomistic data to macroscopic observable (fracture toughness)
- To calculate fracture toughness of wet silica glass from atomistic molecular simulation data
- To investigate the impact of different simple aqueous solution compositions on fracture propagation and toughness:
  - Cations:  $\text{Li}^+$ ,  $\text{Na}^+$ ,  $\text{Ca}^{2+}$
  - Anions:  $\text{OH}^-$ ,  $\text{Cl}^-$
- To compare modeling and experimental results.

# Fracture Mechanics Theory: Fracture Toughness

## Griffith Criterion:

$$\sigma_{cr} = \left( \frac{2E\gamma}{\pi c} \right)^{\frac{1}{2}}$$

**Strain energy release rate** (G) describes the energy stored in material prior to fracture

A linear elastic solution for a slit crack in plane strain mode I leads to:

$$G_{IC} = J_{IC} = \frac{K_{IC}^2(1-\nu)^2}{E} = 2\gamma_s$$

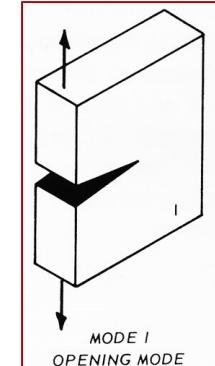
(brittle material)

$$G_{IC} = G_{diss} + 2\gamma_s$$

(material with inelastic behavior)

**Fracture toughness ( $K_{IC}$ ):** the energy required to propagate a crack in a material.

$\sigma_{cr}$  = critical stress  
E = Young's modulus  
c = crack length  
 $\gamma$  = surface energy  
 $K_I$  = stress intensity factor  
 $\nu$  = Poisson's ratio  
G = strain energy release rate



J-integral: a path independent contour integral used to calculate G for monotonic loading

$$J = \int_{\partial\Omega} S \cdot dA$$

Eshelby stress field:

$$S = WI - H^T P$$

I is identity tensor

H = displacement gradient, measure of strain

P = stress

W = energy

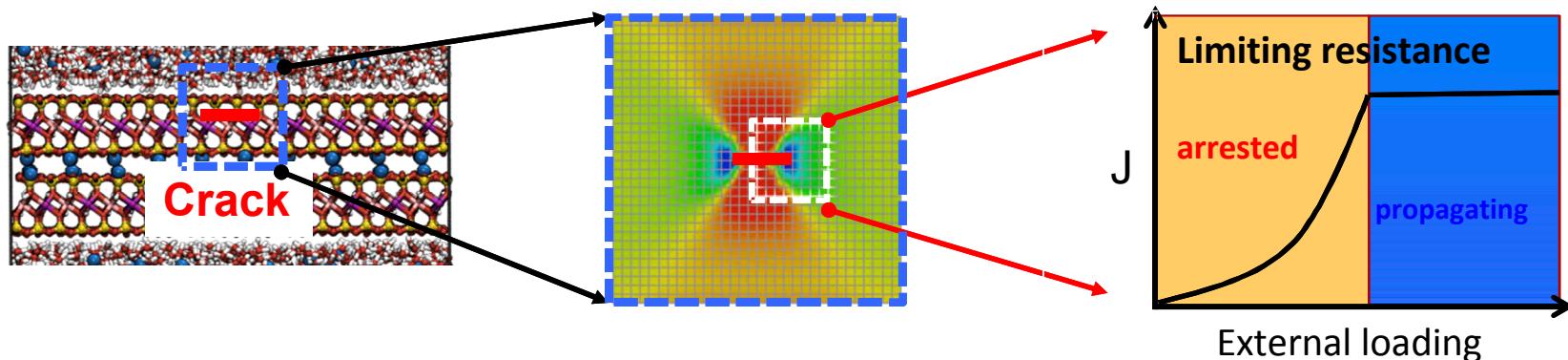
# Mechanical Fracture Models

## Upscaling Metrics: Atomistic to Continuum (AtC)

Molecular simulations sample atomic displacements, energies & forces

Eshelby stress fields use atomic data to characterize local energy available to move the crack (*in red*)

The J-integral is formed from the divergence of the Eshelby stress and determines if a crack will propagate

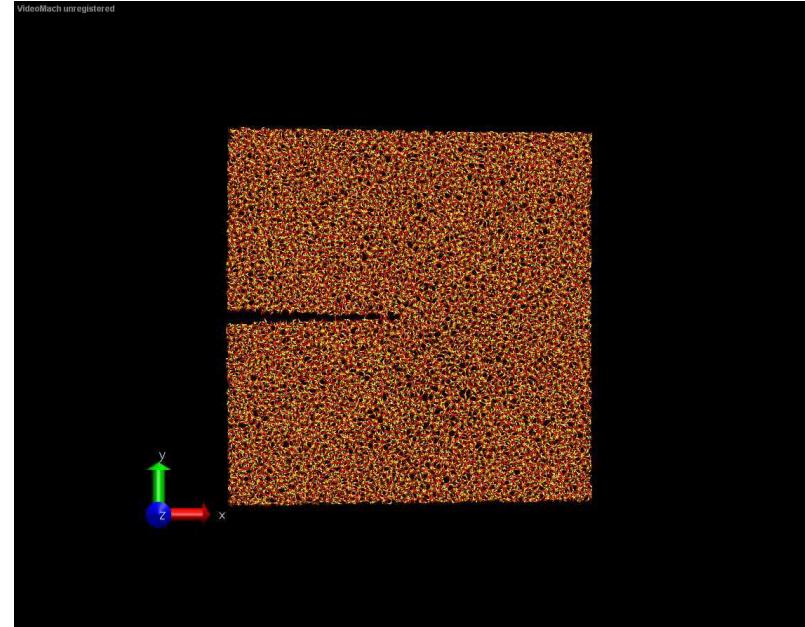
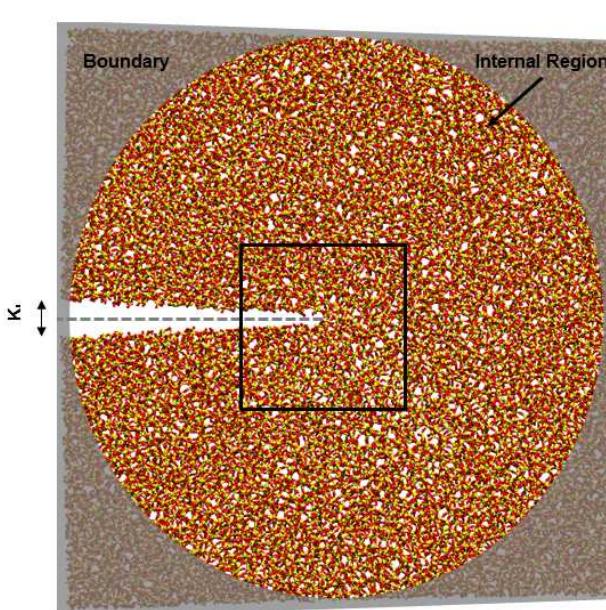


**Jones & Zimmerman: First to successfully apply continuum fracture theory to atomistic systems to estimate fracture toughness**

- The J-integral provides the characteristic crack tip driving force regardless of system configuration.
- **When  $J$  reaches the material's fracture toughness ( $J_c$ ), crack propagation begins.**
- $J/J_c$  can be measured experimentally using geometries and procedures specified in standards (e.g. ASTM E1820 - Standard Test Method for Measurement of Fracture Toughness)

# Simulation Set-Up for Dry Silica System

- ReaxFF Force Field, LAMMPS MD code (Plimpton, XXXX)
- Amorphous silica, system size: 38,400 atoms ( $143 \text{ \AA} \times 143 \text{ \AA} \times 28 \text{ \AA}$ ); Melt and quenched from  $\beta$ -cristobalite system; 12 replicas
- A slit crack is formed creating a singularity with highly concentrated stress
- Boundary atoms are fixed; Far-field loading as Mode I fracture
- Interior atoms are allowed to move freely
- Stress is introduced iteratively by increasing the crack width
- Atomic positions adjust to accommodate the added stress



# Silica Fracture and Energy Dissipation

- Fracture propagates in distinct steps when the stress at the crack tip exceeds the strength of the material
- Perfectly brittle fracture will have no dissipation energy, with all energy used to propagate the fracture
- Dissipation energy calculated from change in energy of the system over the added surface area

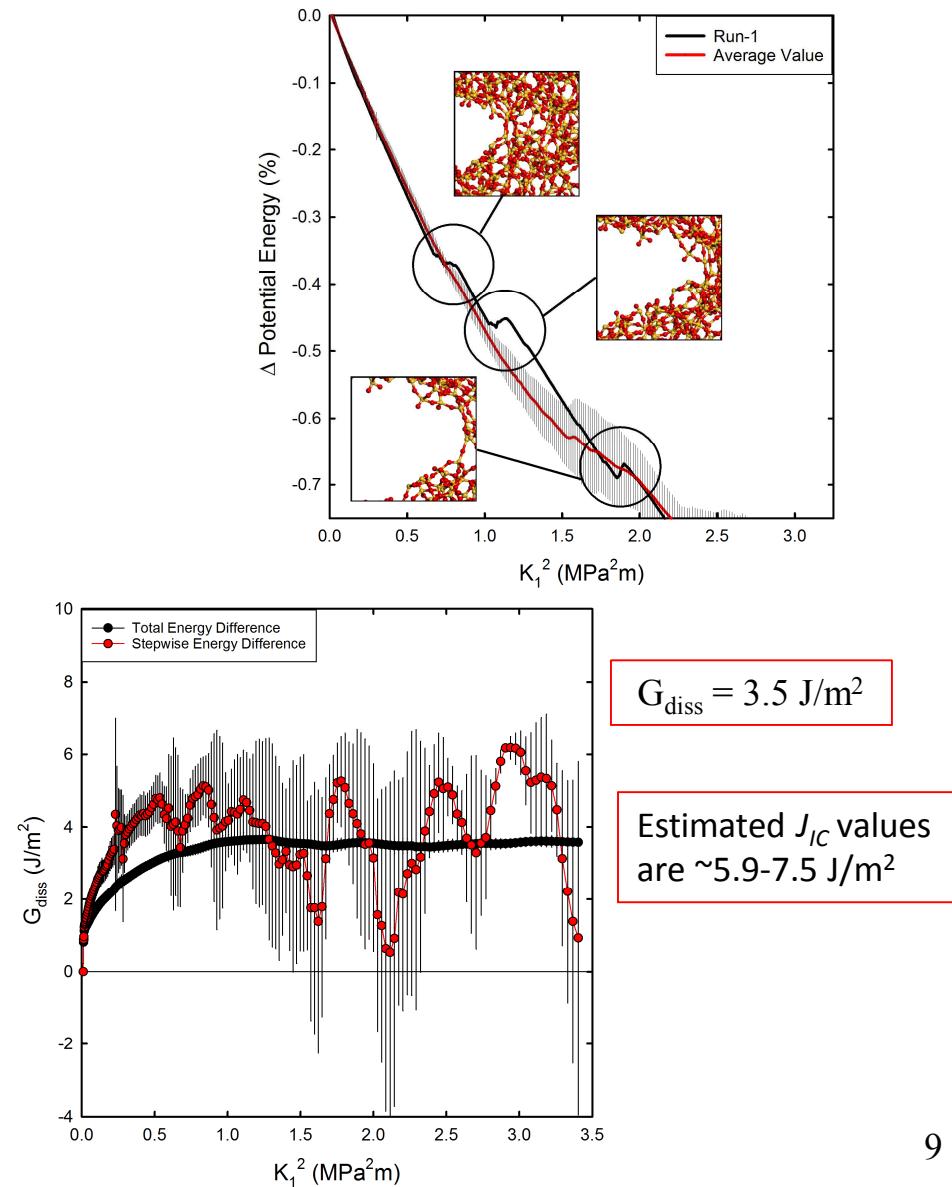
$$\frac{\Delta U}{\Delta S_A} = G_{diss}$$

- Local inelastic behavior as energy is introduced into the silica during loading and not completely dissipated once fracture occurs

$$J_{IC} = G_{IC} = G_{diss} + 2\gamma_s$$

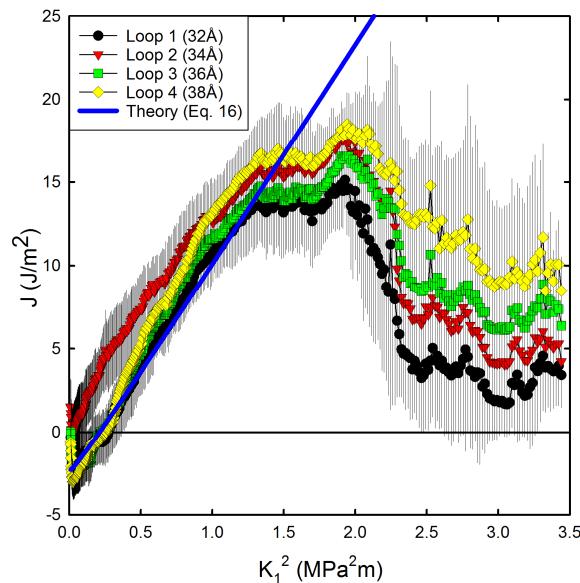
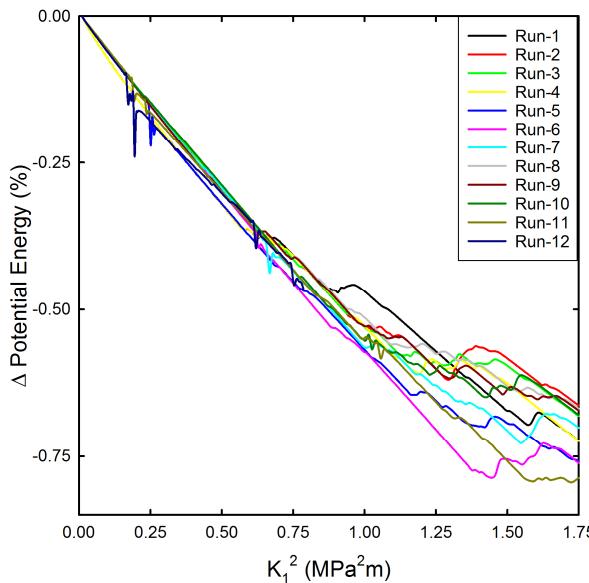
- Surface energies ( $\gamma_s$ ): 1.2-2 J/m<sup>2</sup>

Rimsza, J. M. et al. *Langmuir* (2017).

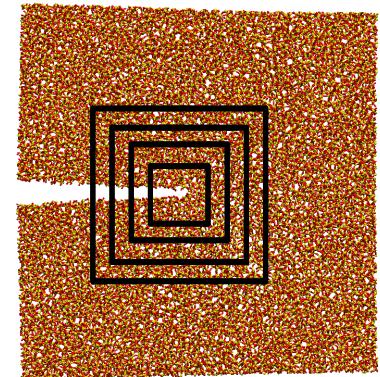


# J-Integral Calculation for Dry Silica

- Calculated via the AtC method through coarse graining energy, displacement, and stress *Jones, Reese E., et al. J. Phys.: Condens. Matter (2010)*
- J-integral converges at loop sizes of  $\sim 3$  nm approximates the size of the inelastic zone



J-integrals with converged loop sizes

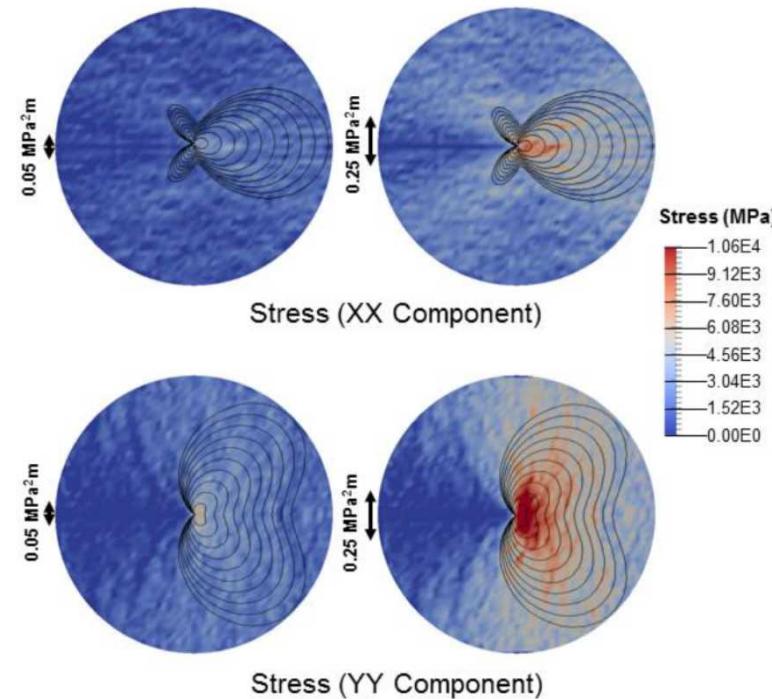
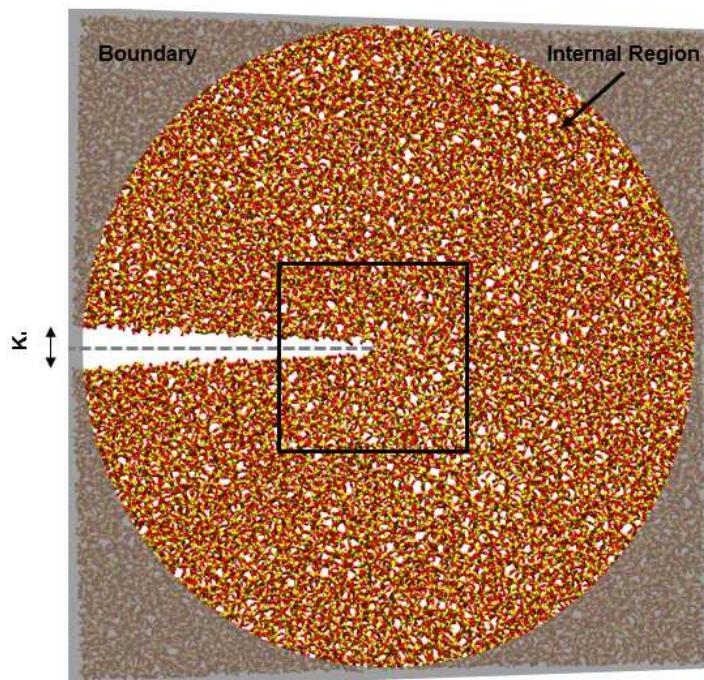


Schematic of increasing loop sizes for J-integral convergence test

Calculated  
 $K_{IC}=0.76\pm0.16$  MPa $\sqrt{m}$   
 $J_{IC}=6.16\pm4.34$  J/m $^2$

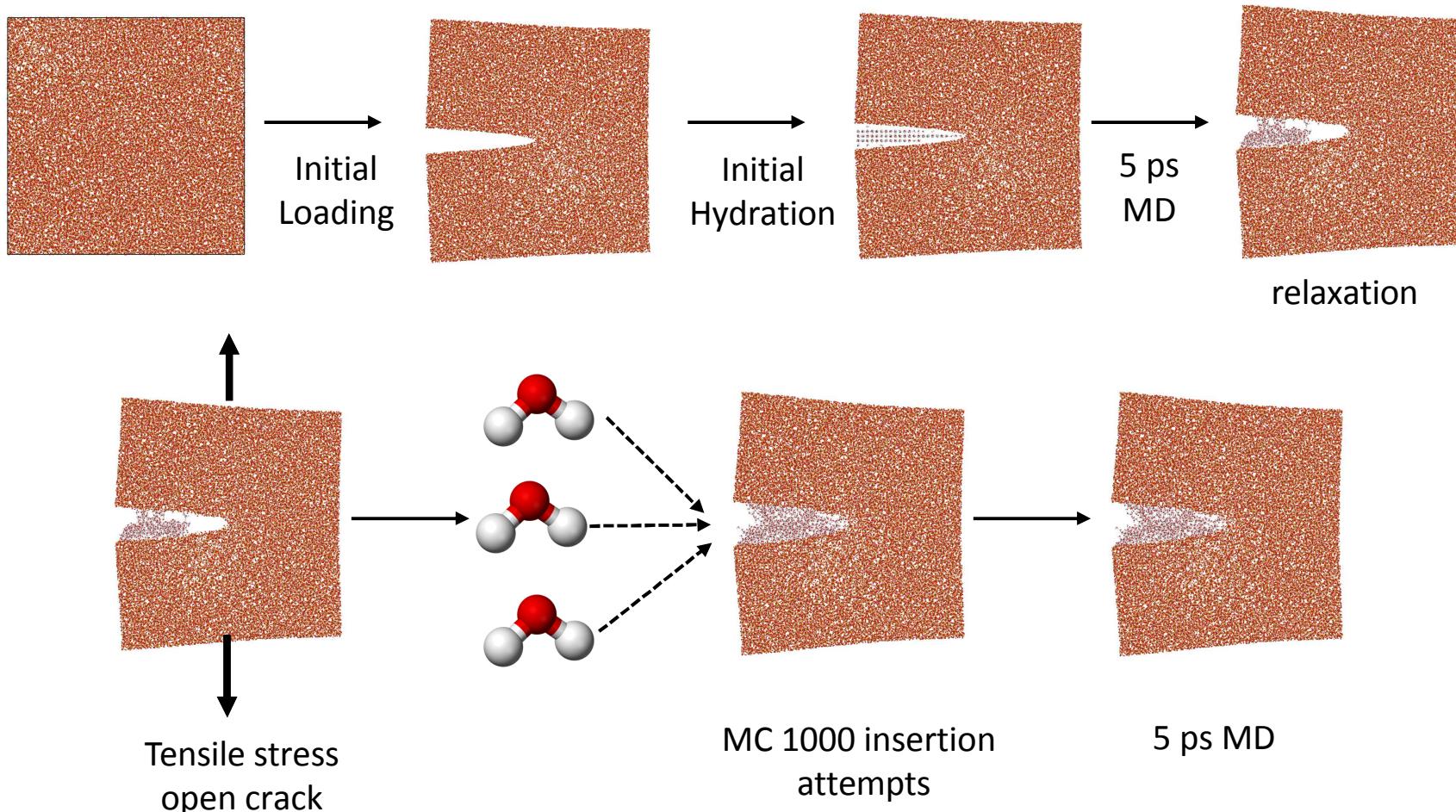
Experimental  
 $K_{IC}=0.78\pm0.04$  MPa $\sqrt{m}$

# Simulated Stress Fields for Dry Silica



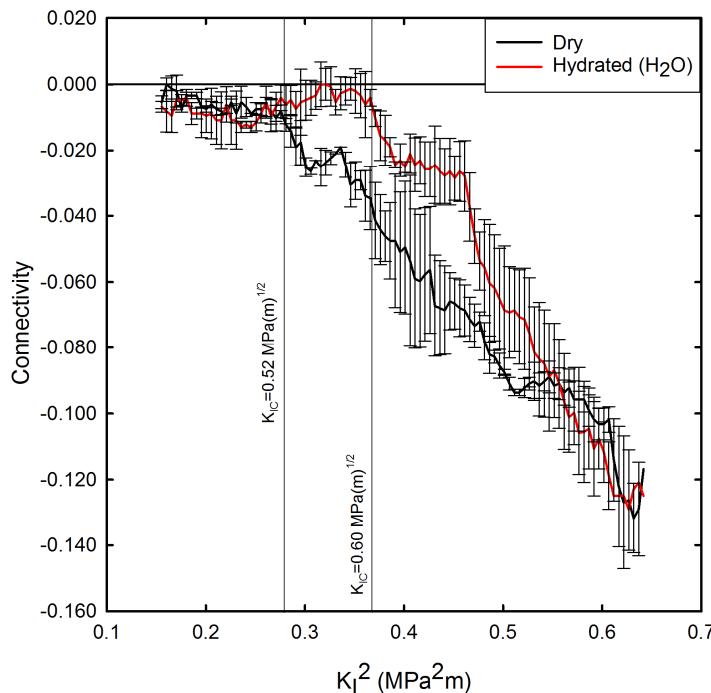
Crack tip stress fields in amorphous silica  
at two loading levels ( $K_I^2 \sim 0.05$  and  $0.25$  MPa<sup>2</sup>m).  
Black contours are from linear elastic solution.

# Silica Fracture in Water: Methodology

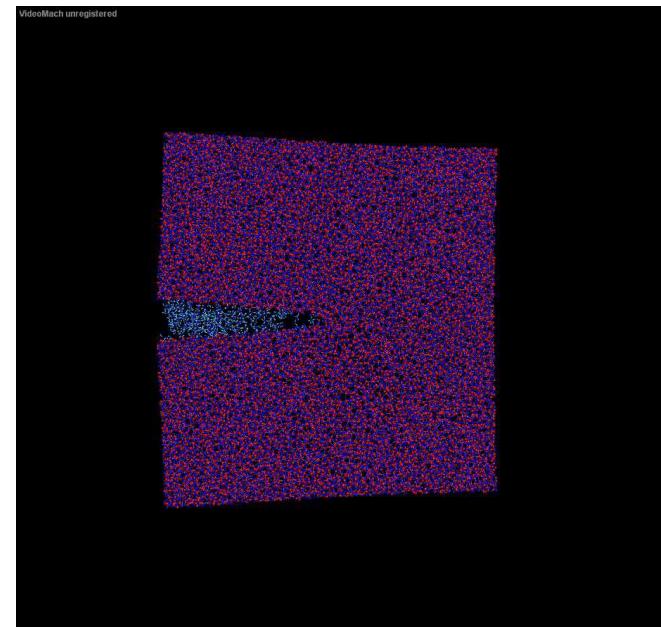


# ReaxFF: Fracture of Hydrated Silica

- Water placed inside the crack followed by far field loading
- Dry silica fractures with less loading ( $\sim 0.52 \text{ MPa}\sqrt{\text{m}}$ ) compared with wet systems ( $\sim 0.60 \text{ MPa}\sqrt{\text{m}}$ )

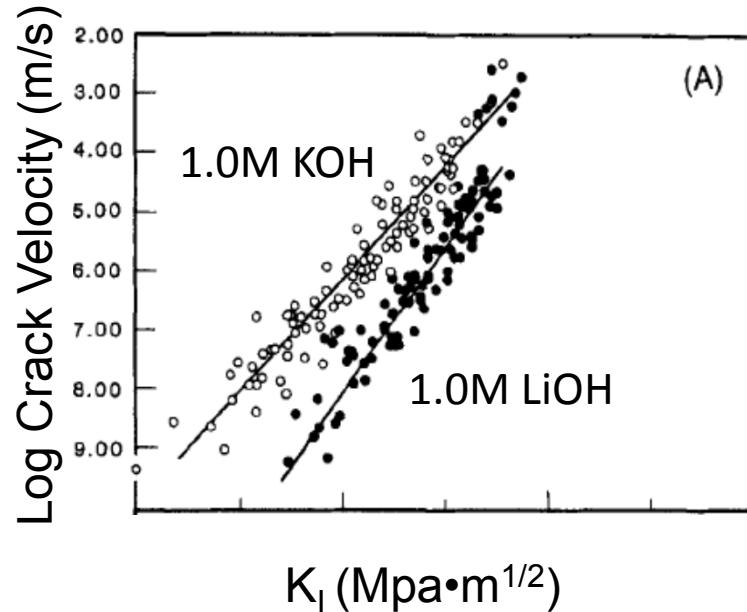
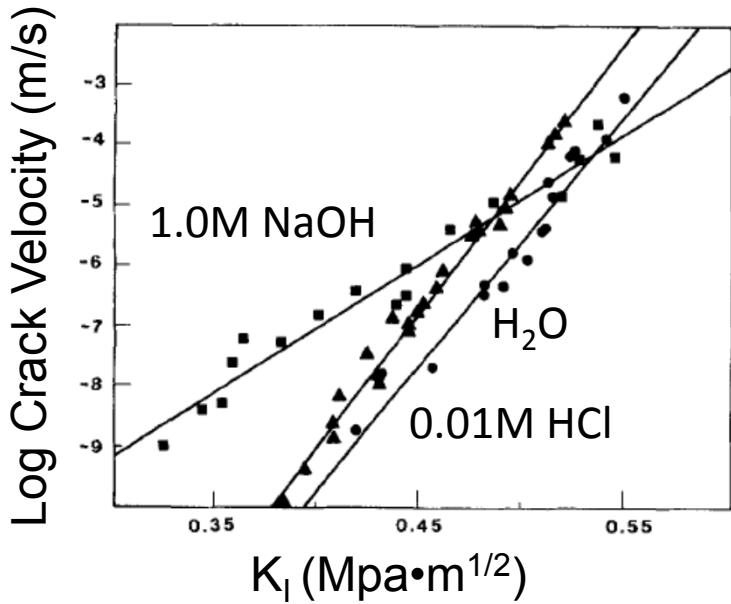


Changing connectivity with loading to identify when a fracture propagation event occurs in dry and hydrated systems



Fracture of hydrated silica  
Colors: water (blue), silica (pink and purple)

# Effect of Counterions on Crack Growth in Vitreous Silica



- The chemical activity of hydroxide is affected by the nature of the cation due to ion association.
- Hydroxide can penetrate the solvation shell of molecules.
- LiOH association decreases OH<sup>-</sup> activity in solution more than NaOH association.
- Dependence of applied stress on cation identity:  
 $\text{NaOH} \sim \text{KOH} \sim \text{CsOH} > \text{LiOH}$

0.1M NaOH     $\text{pK} = -0.7$   
0.1M LiOH     $\text{pK} = -0.08$

1M NaOH    9% Associated  
1 M LiOH    26% Associated

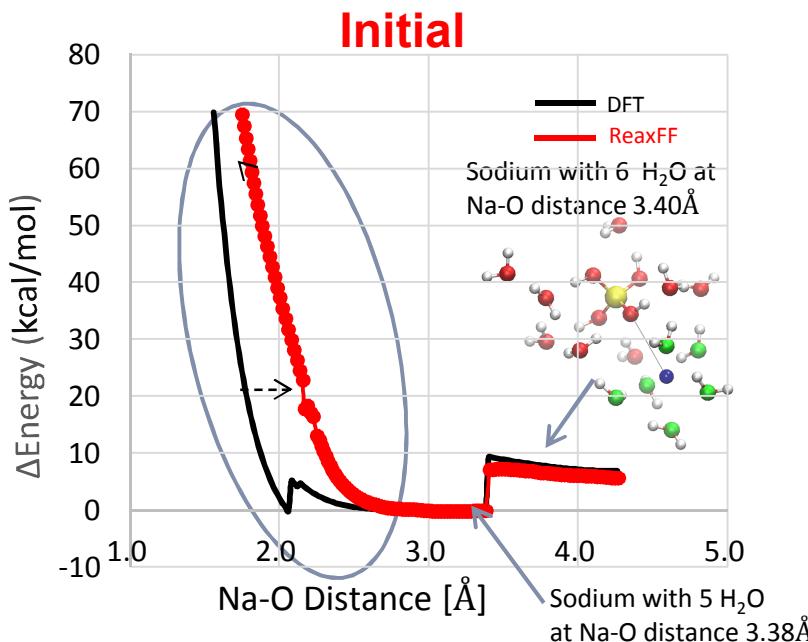
White *et al.*, 1987

# REAXFF Development

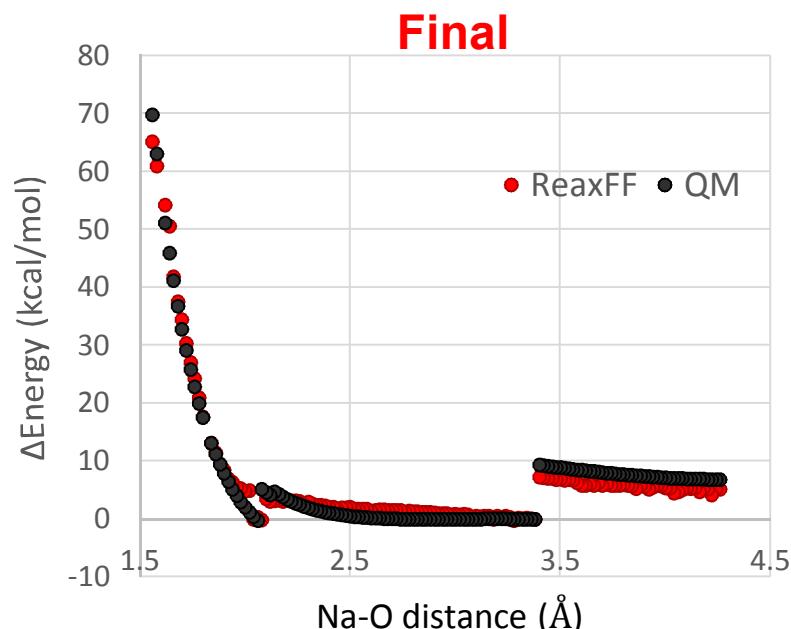
Adri van Duin, Penn State



- Parameters for aqueous electrolytes under development at PennState
- NaOH, KOH, LiOH: Exhibits differences in association constants that are observed experimentally
- Collaboration between PennState and Sandia to develop parameters for  $\text{Na}^+ \text{-H}_2\text{O}$ - $\text{SiO}_2$ ,  $\text{Li}^+ \text{-H}_2\text{O}$ - $\text{SiO}_2$

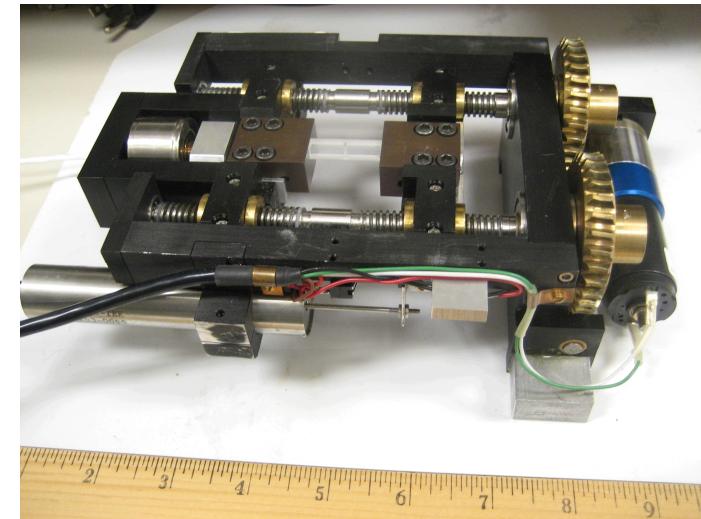
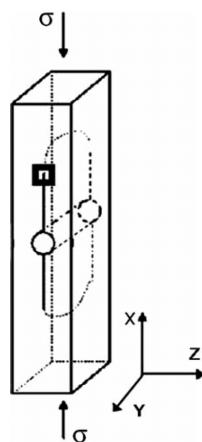
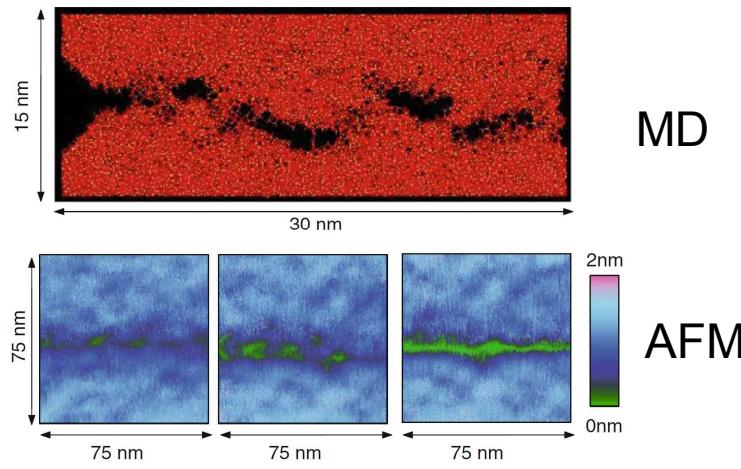


- Good agreement in 2.5-4.5 Å range
- ReaxFF is too repulsive below 2.5 Å



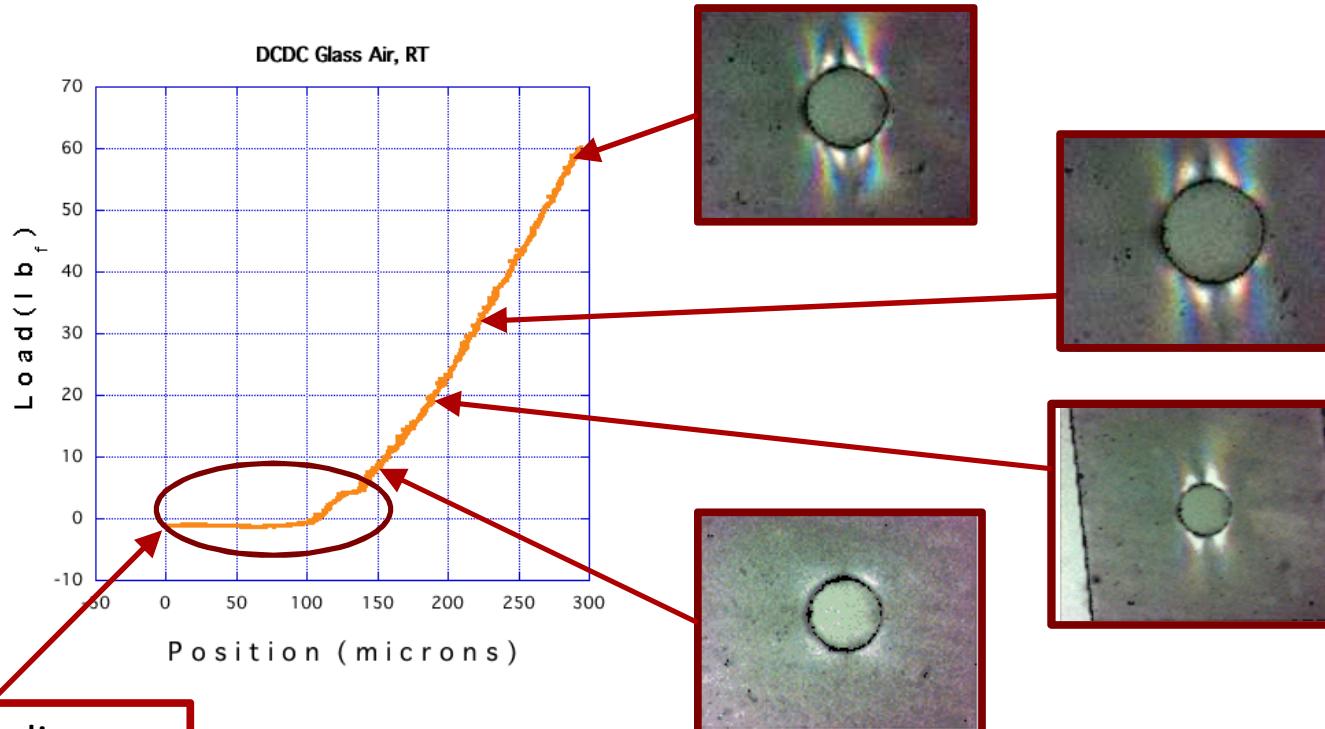
- Excellent agreement through the entire range, including energy jumps around 3.38 Å and 2.1 Å associated with H-bond configuration changes

# Experimental Validation of $K_{IC}$



Review articles: Ciccotti, J. Phys. D 2009, Ciccotti et al. JNCS 2008

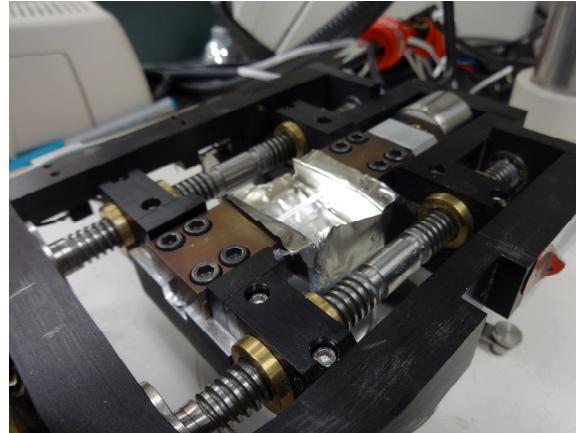
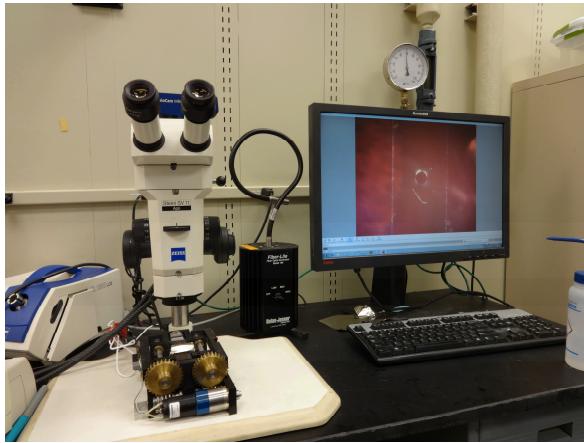
# Imaging Capabilities and Preliminary Results for Dry Silica



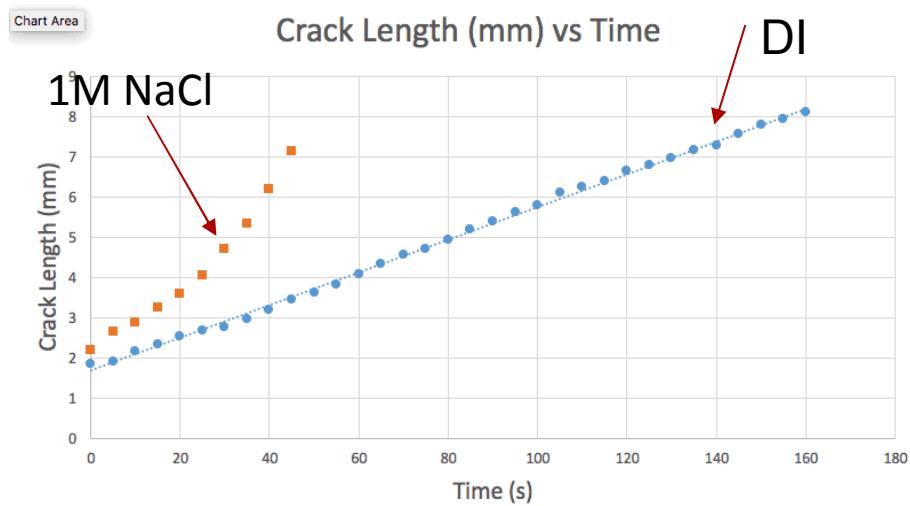
Sample loading  
and compliance of  
nickel foil

- Optical profilometry for post-test analysis and preliminary imaging of load frame tests of DCDC samples
- Polarized light microscopy for visualizing stress fields; will use Moiré analytical methods for stress field quantification

# Experimental Results in Solution



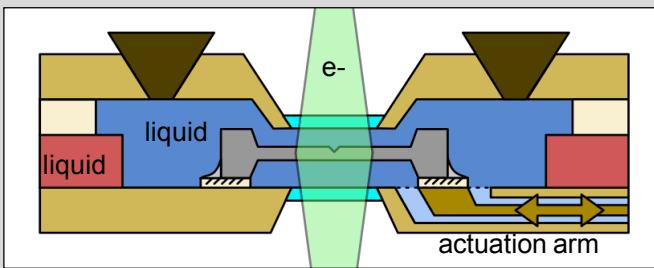
- Tested design for loading in liquids
- Repeatable loading of DCDC glass samples in DI water
- Air, DI, and 0.5 NaCl tested
- Upgraded test equipment for load frame and imaging
  - Software, updated laptop (Windows 10)
  - Time lapse software
  - Upgraded USB microscope



# Different SNL Project Developing Liquid-Mechanical TEM Platform

- First-of-its-kind MEMS device for quantitative testing the mechanical properties of materials within a controlled (chemical and temperature) liquid environment that is coupled with high-resolution real-time imaging
- Through BES support at CINT, this capability will be available to the international research community

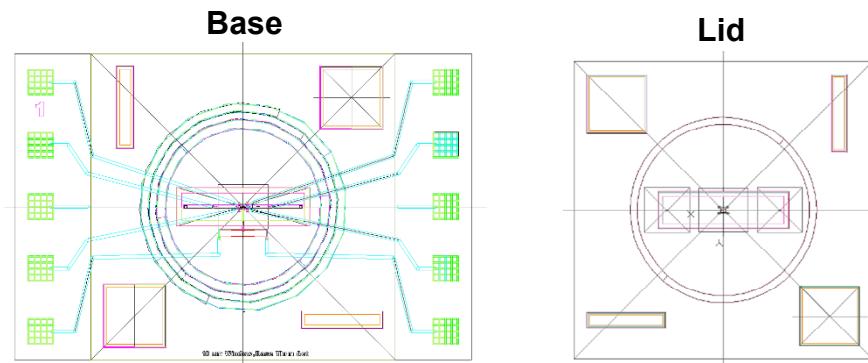
## Generation 1 Concept



- Buried movable arm attached to sample
- Sample suspended over large pads
- Heating of liquid seal-ring
- Prevention of sample/window stiction
- SAM in movable arm and liquid seal-ring
- Low and high force actuators

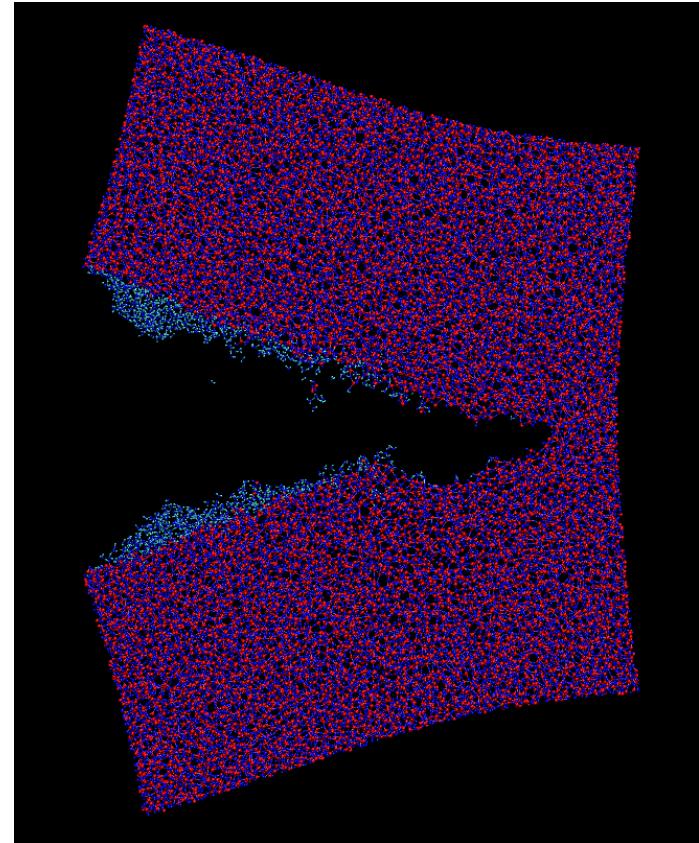
## Liquid-Mechanical MEMS Device

- Nano Fracking TEM Base: 15 modules, 11 different designs, 240 usable die per wafer
  - 11 designs for testing of heaters, window dimensions, window thickness, sample placement, actuation strength, single vs. dual actuation, and capacitive sensing
- Nano Fracking TEM Lid: 25 modules, 4 different designs, 400 usable die per wafer
  - 4 different designs based on window dimensions and thickness



# Summary and Conclusions

- First time atomistic simulation data has been used to calculate the macroscopic J-integral and crack stress fields from atomistic data for an oxide.
- First time ReaxFF has been used to investigate subcritical crack propagation using a slit crack and Mode I tension.
- First time that a molecular-scale simulation method has been developed for stress-corrosion cracking in DI water.
- First time that DCDC experiments have been conducted in environmental chambers – new experimental design
  
- In simulations for dry silica, a localized high stress inelastic region develops prior to fracture with an estimated radius between 3-3.2 nm.
- The calculated  $K_{IC}$  of  $\sim 0.76$  MPa $\sqrt{m}$  is consistent with experimental data reported in the literature.
- Preliminary data is consistent with conceptual models and previous experimental data.



Snapshot of fracture silica in a hydrated environment

## Acknowledgements

*This work was fully supported by the Laboratory Directed Research and Development (LDRD) program of Sandia National Laboratories.*