



Methods for Atomistic Simulations of Soft Matter

Amalie L. Frischknecht
Sandia National Laboratories

GSOFT Short Course
APS March Meeting
March 12, 2017



Sandia National Laboratories is a multi-program laboratory managed and operated by Sandia Corporation, a wholly owned subsidiary of Lockheed Martin Corporation, for the U.S. Department of Energy's National Nuclear Security Administration under contract DE-AC04-94AL85000.

Acknowledgments

- Gary Grest, Mark Stevens, Steve Plimpton (Sandia)
- Pieter in 't Veld, BASF

- Dan Bolintinenu (Sandia)
- Robert Middleton, Karen Winey (Penn)
- Chris Soles, Jacob Tarver (NIST)

Funding: Sandia LDRD
Center for Integrated Nanotechnologies



Why do simulations?

Daan Frenkel: 'In the past, we had to think about the role of simulations because they were expensive, now we have to think because they are (mostly) cheap.'

D. Frenkel, *Simulations: The dark side*, *Eur. Phys. J. Plus* 128, 10 (2013)

Outline

- why atomistic MD?
- force fields
- practical matters
 - codes
 - building starting states
 - timesteps, thermostats, etc
- properties
- what can we learn? some examples...

Why atomistic MD simulations?

Fundamental insight

- accurate molecular model for a particular material
 - structure-property relationships
 - direct comparisons with experiment
- molecular level information not obtainable from experiment
- new insights into material properties from synergistic work between experimentalists and simulators

Property predictions

- novel materials not yet synthesized
- materials that are difficult to measure
- molecular design

Atomistic MD of soft matter

What kinds of systems do people study?

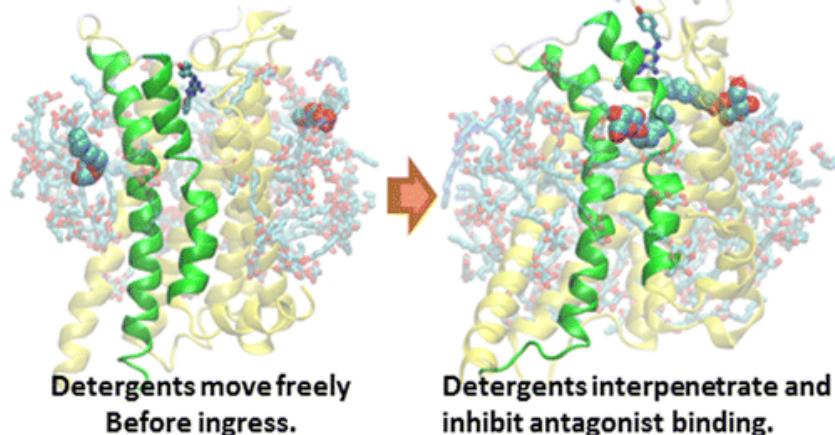
- biomolecules: proteins, peptides, DNA, RNA, lipids,
- polymers (often really oligomers)
- surfactants
- liquid crystals
- nanoparticles (often with attached ligands)

Properties/Phenomena

- morphology
- phase behavior
- self-assembly
- dynamics: transport, viscosity, structural transitions, ...

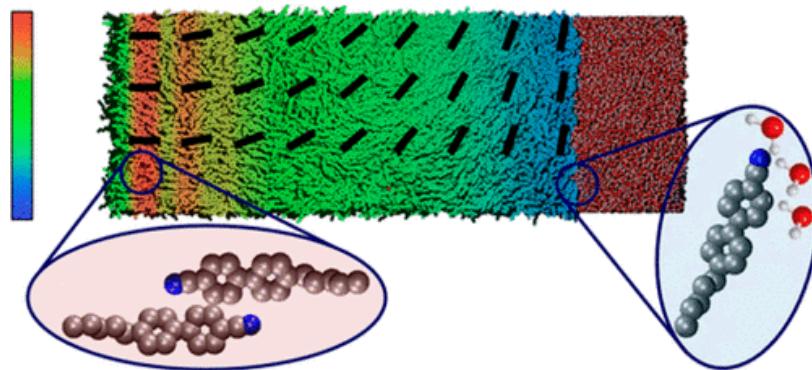
Atomistic MD of soft matter

Tumbling and Penetration of shorter chain detergent



detergent stabilization of G-protein coupled receptors

Lee, S., Mao, A., Bhattacharya, S., Robertson, N., Grisshammer, R., Tate, C. G., & Vaidehi, N. (2016). *JACS*, 138, 15425–15433 <http://doi.org/10.1021/jacs.6b08742>

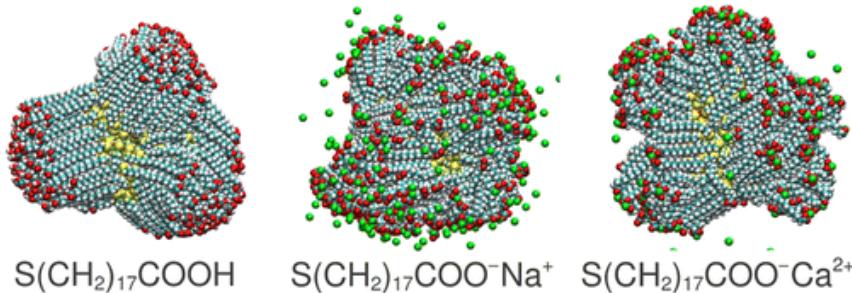


liquid crystals at interfaces

Ramezani-Dakhel, H., Sadati, M., Rahimi, M., Ramírez-Hernández, A., Roux, B., & De Pablo, J. J. (2017). *J Chem Theory Comp*, 13, 237–244.

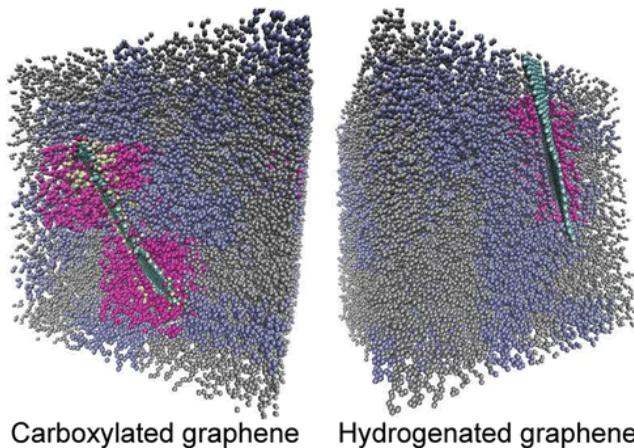
Atomistic MD of soft matter

4 nm Gold Nanoparticles in Water



gold nanoparticles with different ligands in water

Bolintineanu, D. S., Lane, J. M. D., & Grest, G. S. (2014).
Langmuir, 30, 11075–11085.



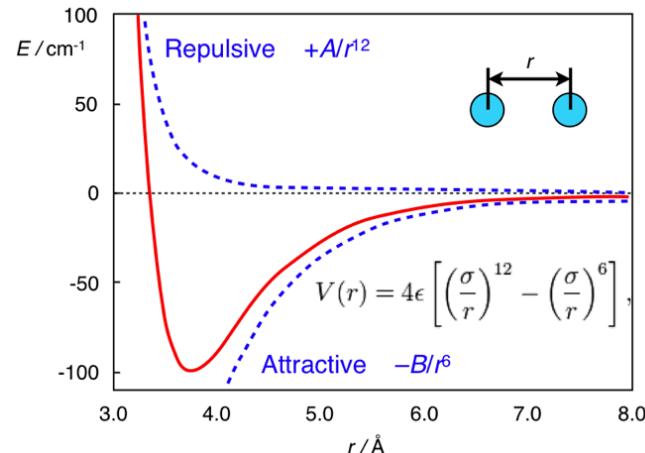
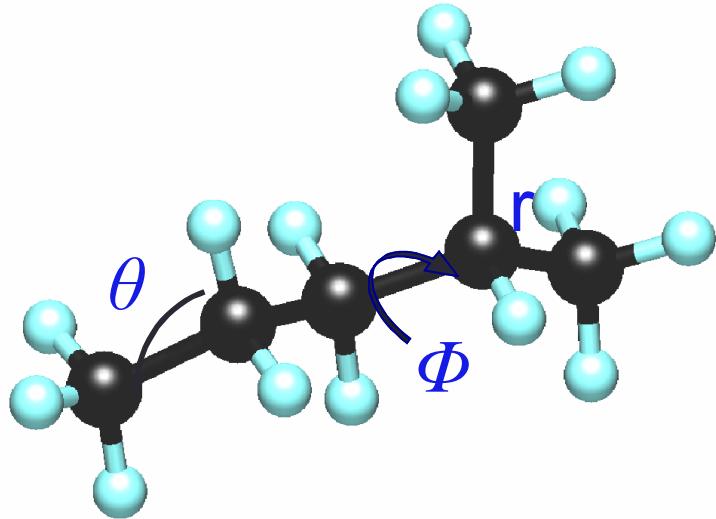
graphene sheets in PE or PEO

Bačová, P., Rissanou, A. N., & Harmandaris, V. (2015).
Macromolecules, 48(24), 9024–9038.

Force fields

interaction potential (energies) between atoms

- intramolecular
 - bonds
 - angles
 - torsions
- intermolecular
 - nonbonded, i.e. van der Waals
 - electrostatics (Coulomb)
 - polarizability



Typical form of an atomistic ff

example: OPLS-AA (all-atom optimized potentials for liquid simulations)

Jorgensen, W., Maxwell, D., & TiradoRives, J. (1996). *J Amer Chem Soc*, 118, 11225–11236.

$$E_{bond} = \sum_{bonds} K_b (r - r_0)^2$$

$$E_{angle} = \sum_{angles} K_\theta (\theta - \theta_0)^2$$

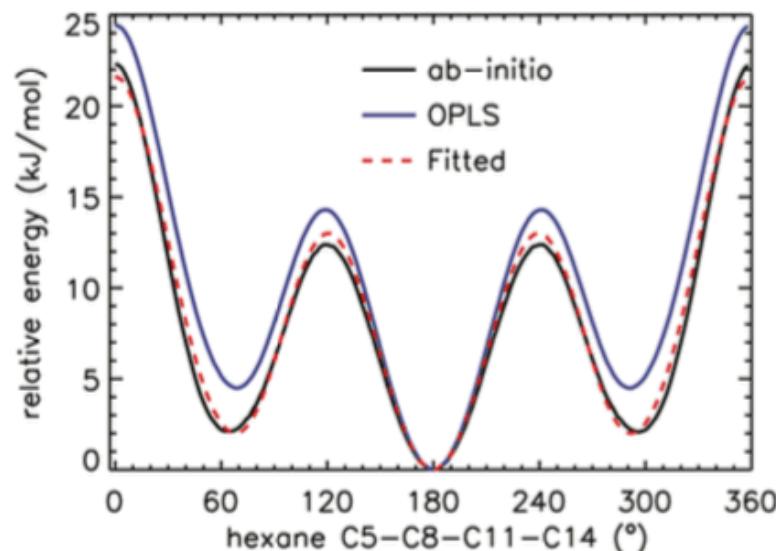
$$E_{torsion} = \sum_i \frac{1}{2} K_1^i [1 + \cos(\phi_i)] + \frac{1}{2} K_2^i [1 - \cos(2\phi_i)] + \frac{1}{2} K_3^i [1 + \cos(3\phi_i)] + \frac{1}{2} K_4^i [1 - \cos(4\phi_i)]$$

$$E_{ij} = \sum_i \sum_j \left[\frac{q_i q_j e^2}{r_{ij}} + 4\epsilon_{ij} \left(\frac{\sigma_{ij}^{12}}{r_{ij}^{12}} - \frac{\sigma_{ij}^6}{r_{ij}^6} \right) \right]$$

this is a lot of parameters!

Developing force fields

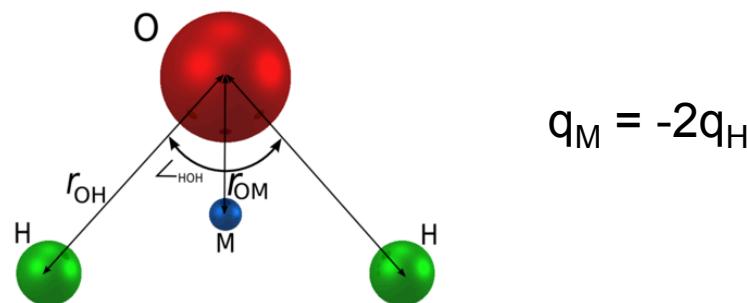
- bonded interactions:
 - bond stretch, angles: from spectroscopy of small molecules or ab initio
- torsions usually fit from *ab initio* (quantum chemistry) calculations
 - calculate energies for a suite of small molecules (gas phase)
 - fit to the functional form of the ff



Developing force fields

- nonbonded interactions
 - often fit to experiment
 - run MD or MC simulation of several (many) molecules
 - compare to: densities, heats of vaporization, hydration energies, ...
 - iterate parameters to get agreement
- partial charges
 - calculate electrostatic potential from quantum chemistry
 - fit partial charges at atom centers

exception: water models often have charge not on an atom, e.g. TIP4P:



More complexity in force fields

class II force fields

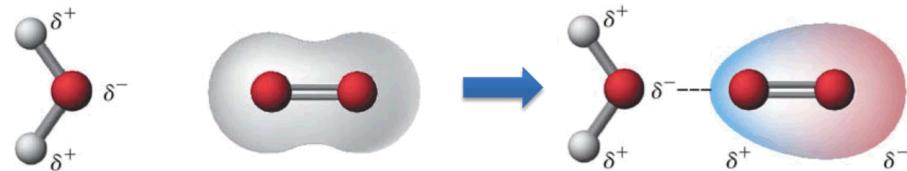
anharmonic terms like $H_3(\theta - \theta_0)^3$

cross-terms like $F_{r\theta}(r - r_0)(\theta - \theta_0)$

polarizability

in reality, charge is not fixed—should respond to environment
include induced dipoles on atoms

$$U_{pol}(\mathbf{r}) = -\frac{1}{2} \sum_i \mu_i \cdot \mathbf{E}_i$$



from Lee-Ping Wang, Stanford

chemical reactions

allow bond formation/breaking

e.g. ReaxFF force field

Common force fields for soft matter

- OPLS-AA (Optimized Potentials for Liquid Simulations-All Atom)
 - Jorgensen group (Yale)
 - transferable; optimized in liquid state
- AMBER (Assisted Model Building with Energy Refinement)
 - originally for biomolecules, Kollman group (UCSF)
 - almost same functional form as OPLS
 - also an MD code
 - <http://www.quimica.urv.es/~bo/MOLMOD/General/Forcefields/AMBER.html>
- CHARMM (Chemistry at HARvard Macromolecular Mechanics)
 - www.charmm.org; led by Martin Karplus
 - also an MD code; free to academic users
 - for biomolecules

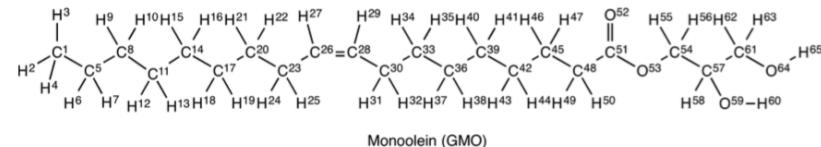
less common:

- COMPASS
 - a class II force field
 - available from Accelrys (not all parameters are published)
- DREIDING
 - Goddard group (CalTech)

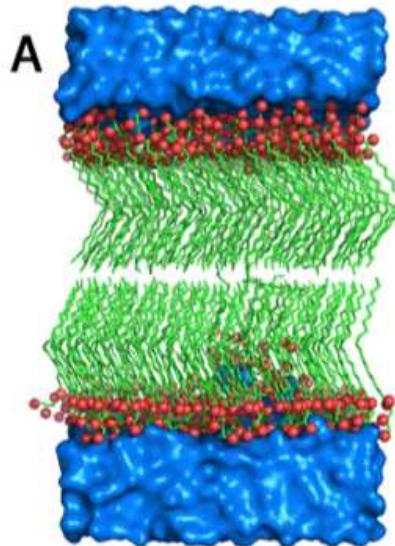
Force field strongly affects properties

example: LOPLS-AA

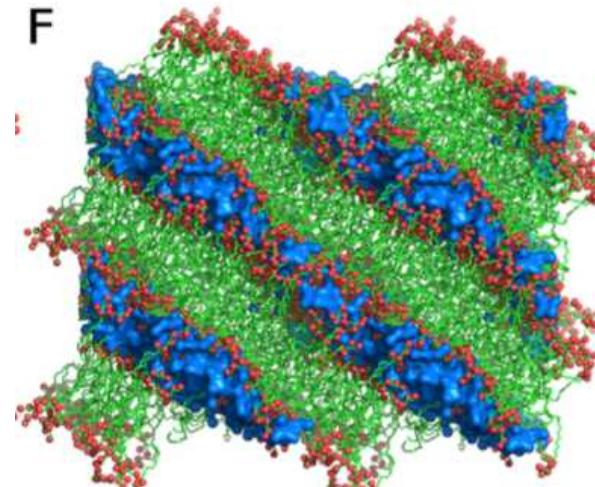
refined to better treat long hydrocarbons
corrects phase behavior of surfactants



incorrect gel phase with OPLS

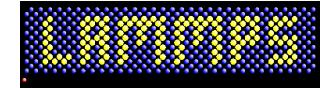


correct liquid bilayer phase with LOPLS



Open source MD codes

- LAMMPS (Large-scale Atomic/Molecular Massively Parallel Simulator)
 - lammps.sandia.gov
 - general code materials science code
 - includes OPLS, CHARMM, Amber ffs
- NAMD (Not Another Molecular Dynamics program)
 - <http://www.ks.uiuc.edu/Research/namd/>
 - designed for biomolecular systems
 - scales to very large systems
 - integrated with VMD
- GROMACS
 - www.gromacs.org
 - primarily designed for biomolecules
 - faster than LAMMPS for some systems



see also:

https://en.wikipedia.org/wiki/Comparison_of_software_for_molecular_mechanics_modeling

Building starting states

need a good starting configuration for simulations

options:

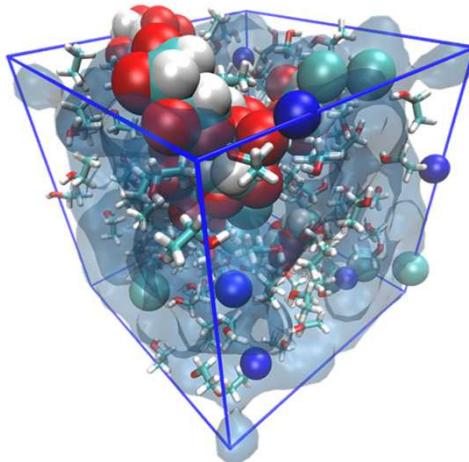
- write your own builder
- use a commercial code like Materials Studio (see: <http://accelrys.com/products/collaborative-science/biovia-materials-studio/>)
- for biomolecules, download a pdb structure
 - e.g. from the Protein Data Bank, <http://www.rcsb.org/pdb/home/home.do>
 - can use tools in VMD to fill with water, etc.
- PACKMOL (<http://www.ime.unicamp.br/%7Emartinez/packmol/home.shtml>)
- build with a Monte Carlo code
 - MCCCS Towhee (<http://towhee.sourceforge.net>)
 - EMC (<http://montecarlo.sourceforge.net>)

Example: building with EMC

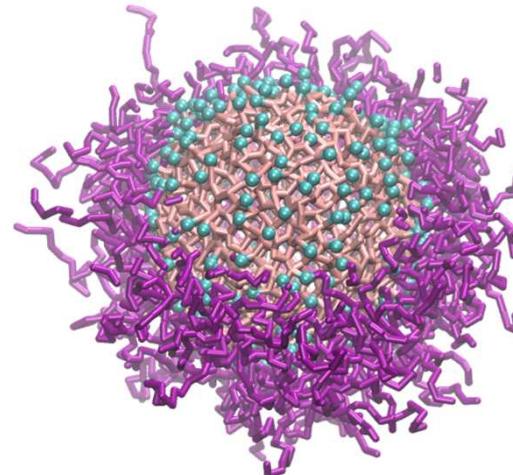
EMC: Found at <http://montecarlo.sourceforge.net/>



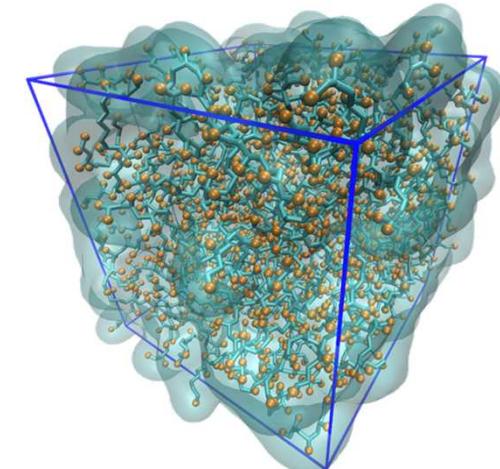
Pieter J. in 't Veld,
ROM/AM, Soft Matter Modeling



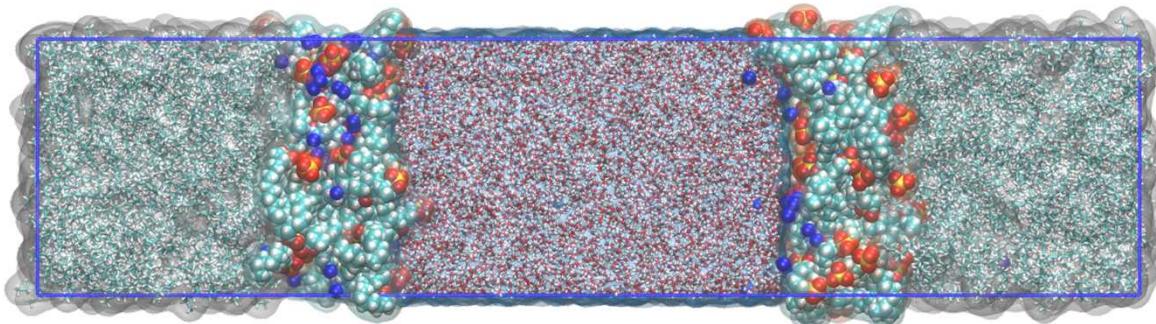
bulk



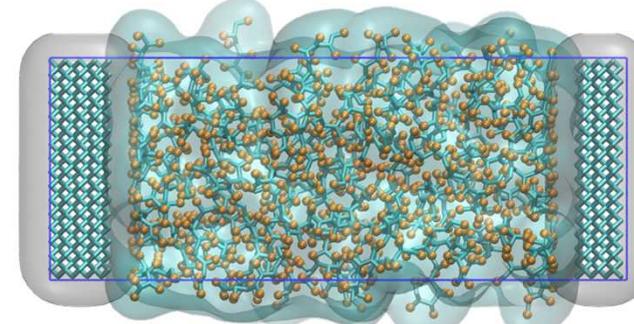
grafting



copolymers



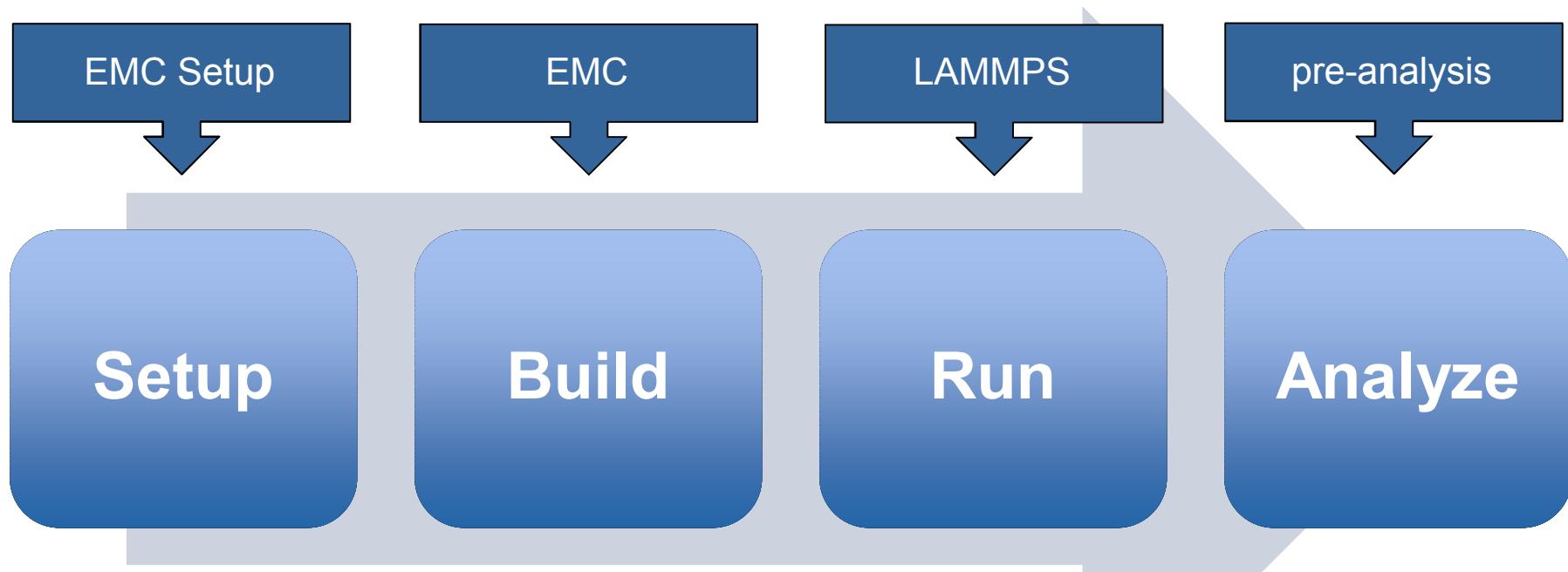
multiphase



surfaces

Workflow for Simulations

Check \$EMC_ROOT/examples/setup



- Force field: PCFF, OPLS, CHARMM, TraPPE
- N , P , ρ , T
- Chemical composition
- Selection of phases
- Supports polymers

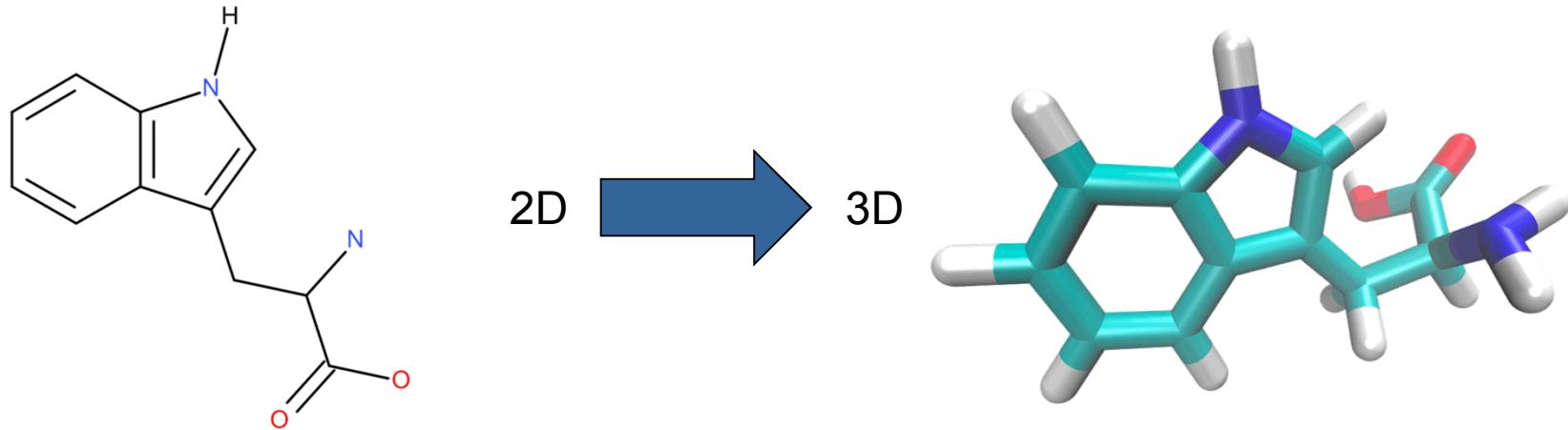
- Input for EMC generated by setup
- Textual format, which can be altered
- Builds condensed phases

- Input for LAMMPS generated by setup
- Holds analysis (e.g. pressure, volume, density profiles etc.)

- Allows for pre-analysis for further data processing

Molecular Structure Investigations

Coupling between 2D and 3D



- Chemical input uses SMILES
- Example: tryptophan (build based on COMPASS force field)
 - Draw structure with Biovia Draw (or JChemPaint (Freeware))
 - Obtain SMILES (both Biovia Draw and JChemPaint convert)
 - Build with EMC (resulting structure dependent on random seed)

Overview

Advantages and Capabilities

- <http://montecarlo.sourceforge.net/>
- Freeware under GPL v3.1
 - Source code in C
 - Command line with scripting interface
 - Wrap-around setup script
- Advantage over commercial solutions
 - Flexible scripting
 - Can build and graft surfaces in condensed media
 - Can build multiphase systems
- Structure builds in
 - Gas phase
 - Condensed phase (also polymers)
- Algorithms
 - Inverse Monte Carlo (CCB growth)
 - Local spatial incremental relaxation
 - Composite core exclusion
- Applications
 - Atomistic and coarse-grained structures
 - Mechanical, rheological and interfacial properties

Overview

Features

- Documentation
- Setup script (`emc_setup.pl`) creates
 - EMC build script
 - LAMMPS input script
- Multiple force fields
 - AA (PCFF, OPLS-AA, CHARMM)
 - UA (OPLS-UA, TraPPE)
 - CG (DPD, SDK, MARTINI, Colloidal)
- Algorithms
 - Inverse Monte Carlo (CCB growth)
 - Local spatial incremental relaxation
 - Composite core exclusion
- EMC build script output
 - LAMMPS data files
 - PDB/PSF and VMD script for use with VMD

Thermostats/barostats

- Nosé-Hoover recommended for atomistic simulations
 - add extra variables to equations of motion
 - couple to particle velocities (control T)
 - couple to system dimensions (control P)
 - has a conserved quantity (extended Hamiltonian)
 - generates the NVT/NPT ensemble
 - ...in the right conditions (center of mass velocity = 0)

can modify to rigorously generate canonical ensemble with Nosé-Hoover chains
this is the default option in LAMMPS!

damping parameters:

too large: large fluctuations

too small: long time to equilibrate

rules of thumb:

temperature damping ~ 100 timesteps

pressure damping ~ 1000 timesteps

Berendsen thermostat: does not generate a canonical ensemble
(used to be default in GROMACS)

Time scale issues

in atomistic simulations, need to resolve high frequency oscillations,
e.g. C-H bonds



time step of 0.1 1 fs

practical time scale of atomistic MD:
ps to 100s of ns (if willing to run a long time...)

heroic MD: μ s

can not parallelize time

Time steps

- For polymers, highest frequency modes typically from bond potential:

$$U_b(r) = k_b(r-r_0)^2/2$$

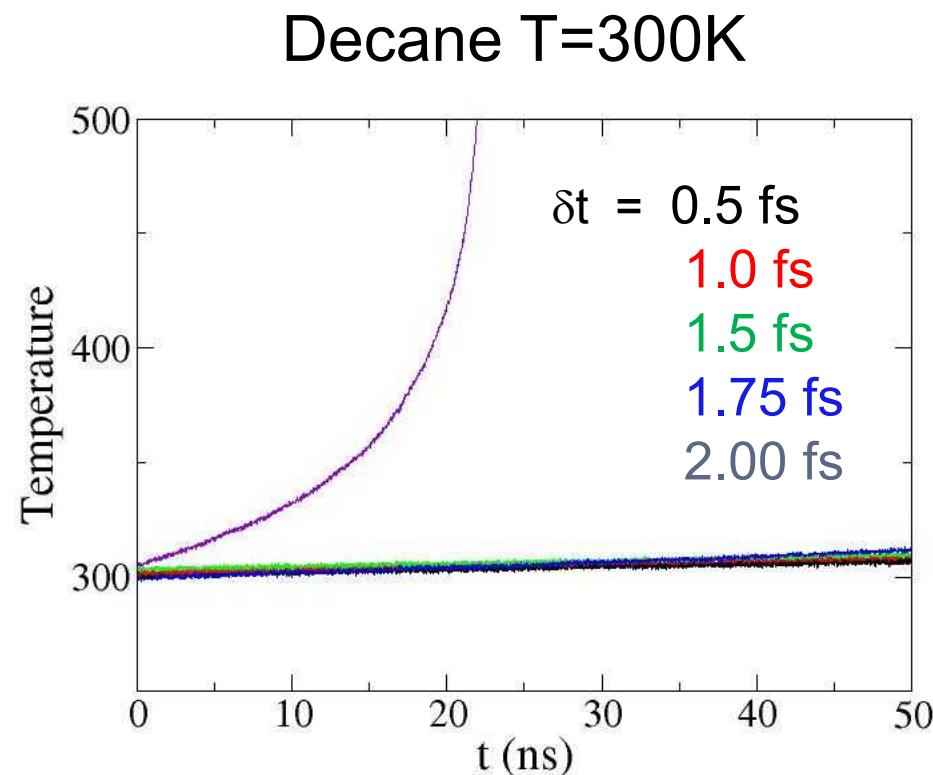
- C-H bond (OPLS)¹

$$k_b = 680 \text{ Kcal mol}^{-1} \text{ \AA}^{-2}$$

- $\omega_b \approx 0.5 \text{ fs}^{-1}$

- $t_b = 2\pi/\omega_b \approx 12 \text{ fs}$

- High temperature, system becomes anharmonic: $\delta t \downarrow$



courtesy of Gary Grest

Ways to accelerate simulations

SHAKE, RATTLE

fix bonds and/or angles
allows larger time steps
often used on H atoms

rRESPA: Reversible Reference System Propagator Algorithms

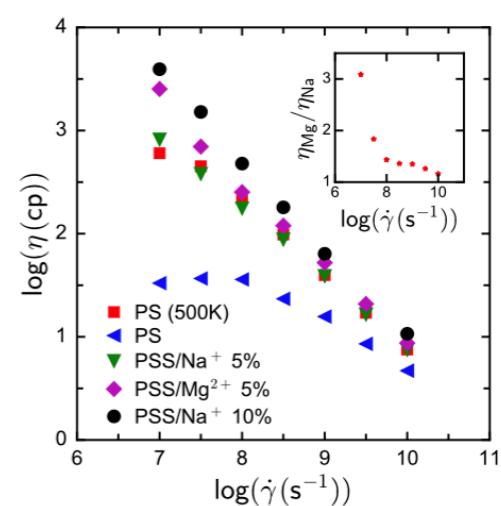
Integrate Bond, Angle, Dihedral, van der Waals (LJ), k-space with different time steps
- LAMMPS – up to 4 time steps

Example: Decane (300K) $\delta t = 1.5$ fs, 38 ns/day, 64 cores (6400 atoms)
rRESPA – $\delta t = 4.5$ fs for k-space, 1.5 fs all other interactions, 48 ns/day

Consequences of time scales

- equilibrium:
 - not practical for e.g. long polymers (reptation time $\sim N^{3.4}$)
 - biomolecules: may have structural transitions on long time scales
- nonequilibrium:
 - imposed deformations often happen too fast
 - shear or strain rates: often unrealistically fast
 - temperature quenches: orders of magnitude faster than experiment
 - difficult to simulate glassy systems
 - NEMD often in nonlinear regime

viscosity of PSS



How do we know if our simulation is accurate?



- compare to experiment
- compare to previous well-established simulation results
- check conservation laws/statistical mechanical relations
 - eg constant energy in NVE ensemble
- calculate the same quantity with two methods
 - example: Green-Kubo expressions vs nonequilibrium simulation

Special techniques

- free energy calculations
 - requires multiple simulations
 - WHAM, thermodynamic integration
- nonequilibrium MD (NEMD)
- parallel tempering
- combined MD-MC
- metadynamics
-

Calculating dynamic properties

- from equilibrium fluctuations
 - i.e. time correlation functions with Green-Kubo formalism
 - often best to calculate in NVE ensemble
- from nonequilibrium simulations
 - be careful with thermostatting

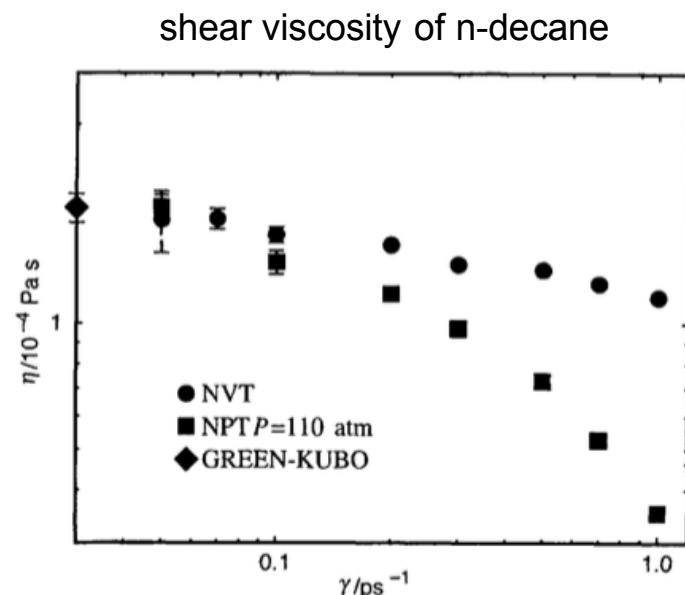


fluctuation-dissipation theorem

example: shear viscosity

Green-Kubo: $\eta = \frac{V}{k_B T} \int_0^\infty \langle P_{xz}(0)P_{xz}(t) \rangle dt$

NEMD: $\eta = -\sigma_{xz}/\dot{\gamma}$



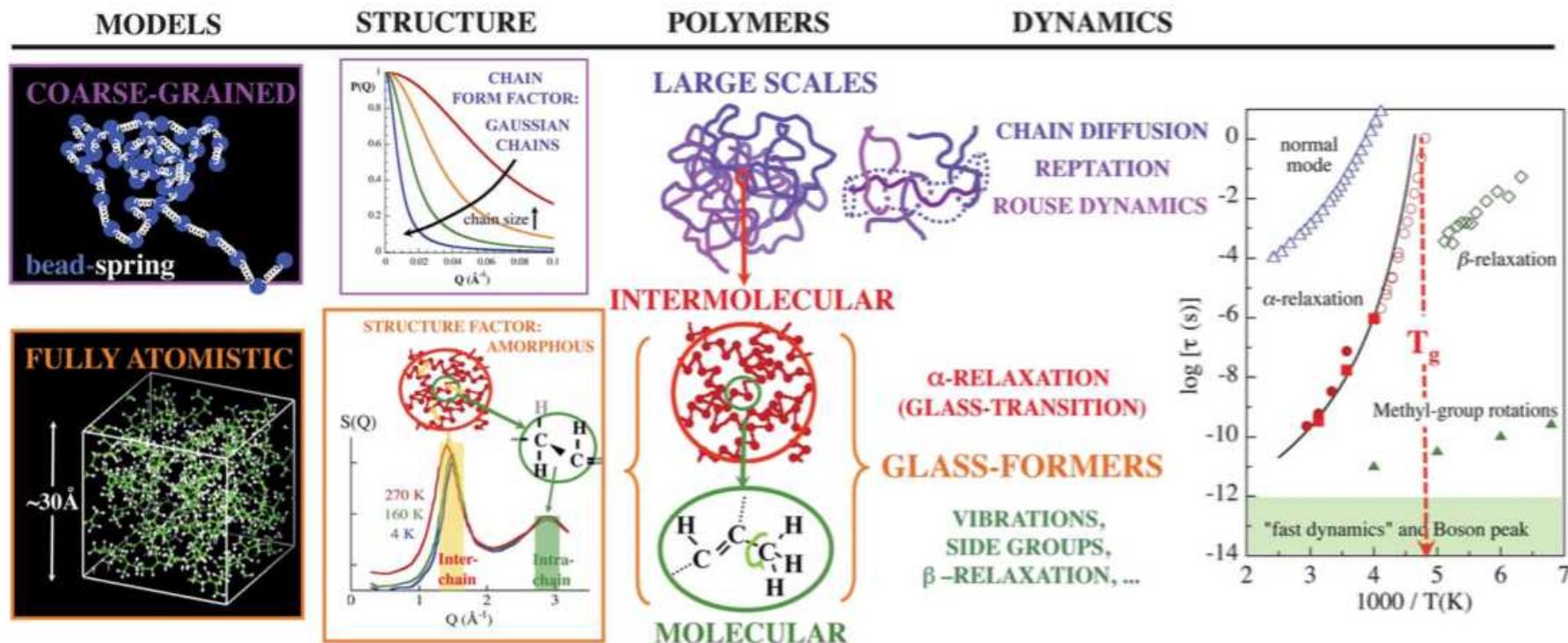
Mundy, C. J., Balasubramanian, S., Bagchi, K., Siepmann, J. I., & Klein, M. L. (1996). *Faraday Discussions*, 104(0), 17–20.

Calculating dynamic properties

- NEMD methods:
 - pro: intuitive to understand, quick to converge
 - con: unphysically large velocity profiles and momentum fluxes
- Green-Kubo method:
 - pro: equilibrium simulation, can use smaller system
 - con: slower to converge
 - con: hard to tell when correlation integral has converged

Atomistic MD and scattering

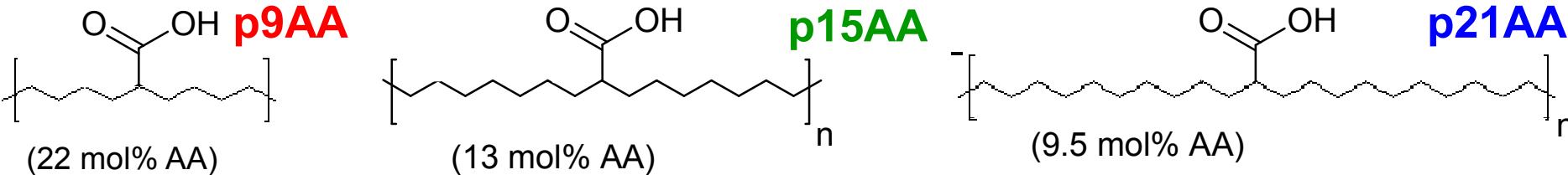
generally overlap well in length and time scales
 many opportunities for collaboration with experimentalists!



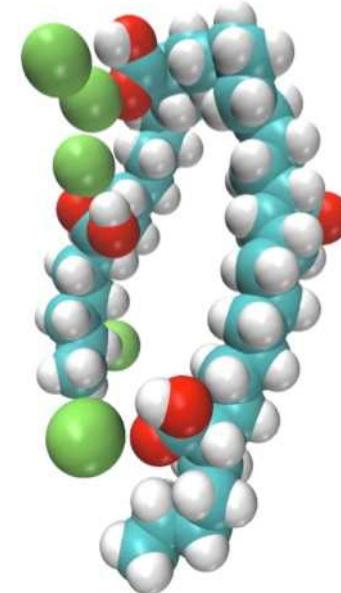
What can we learn?

a personal example: atomistic simulations of ionomers

PE backbone with **precisely** spaced carboxylic acid functional groups



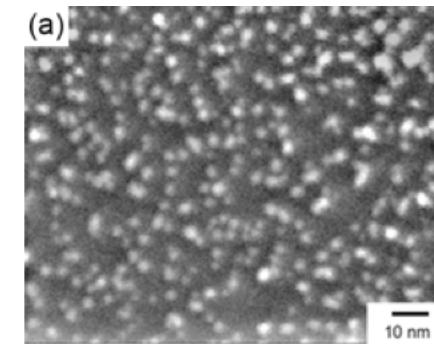
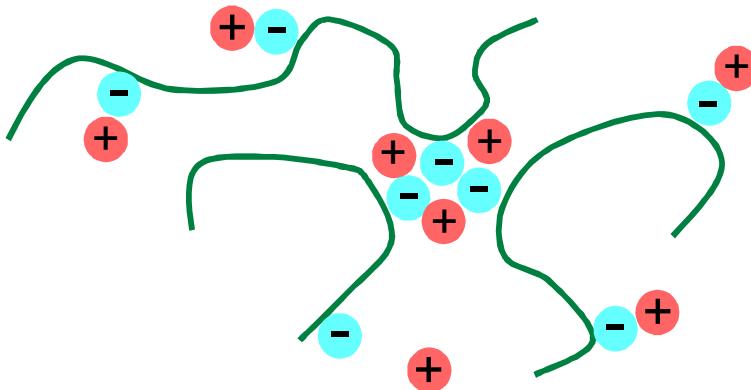
- All atom OPLS-AA force-field
- 80-200 polymers, $n = 4$ repeat units (4 acid groups)
 - $\sim 64 \text{ \AA}$ box, total of $\sim 25,000$ atoms
- NVT ensemble, **150°C** \rightarrow well above T_g
- 30 ns (400 ns in one case)
- LAMMPS



Goals

morphology: understand ion aggregate morphology

Coulombic forces favor aggregates
polymer entropy limits size



ionic aggregates

Seitz et al., J Am Chem Soc (2010) 132, 8165

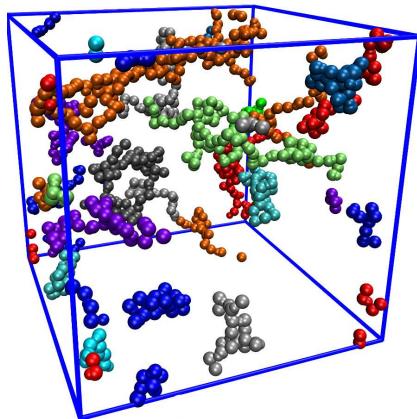
nm-scale ionic aggregates

dynamics: understand how morphology impacts polymer, ion dynamics

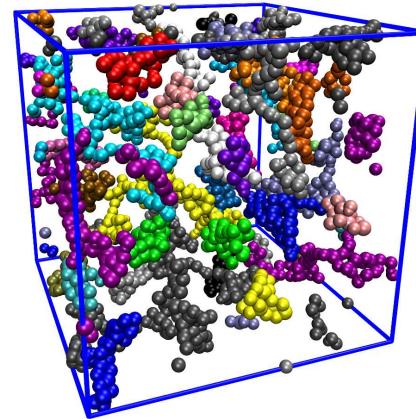
Morphology: Li-neutralized pAA

coloring by cluster

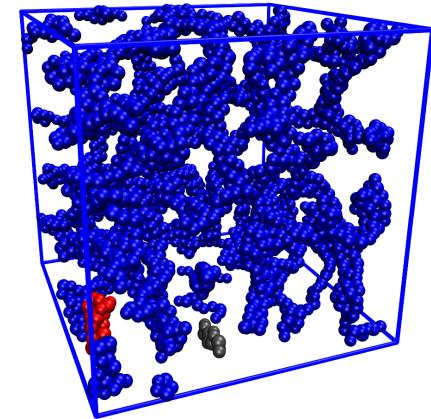
p9AA-10%Li



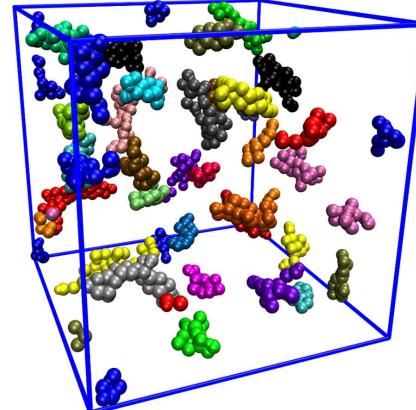
p9AA-43%Li



p9AA-100%Li



p21AA-43%Li



Bolintineanu et al, *ACS Macro Lett*, 2013

Structure Factors from atomistic MD

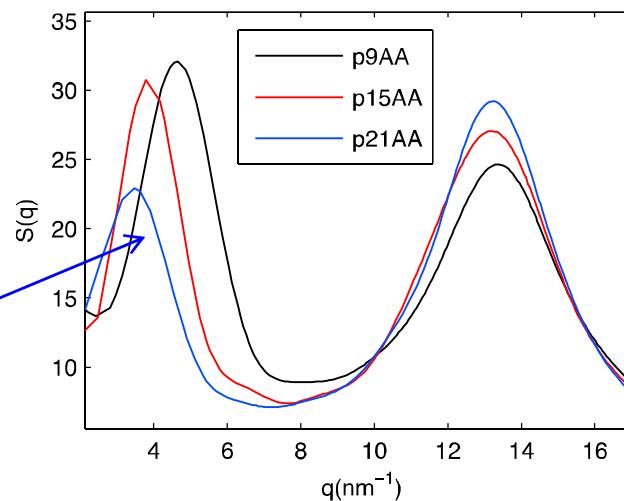
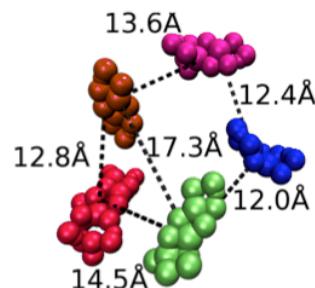
$$S(q) = \sum_i c_i f_i^2 + 4\pi\rho \int_0^\infty \frac{\sin(qr)}{qr} r^2 \sum_{i,j} c_i c_j f_i f_j (g_{ij}(r) - 1) dr$$

c_i = mole fraction of atom i

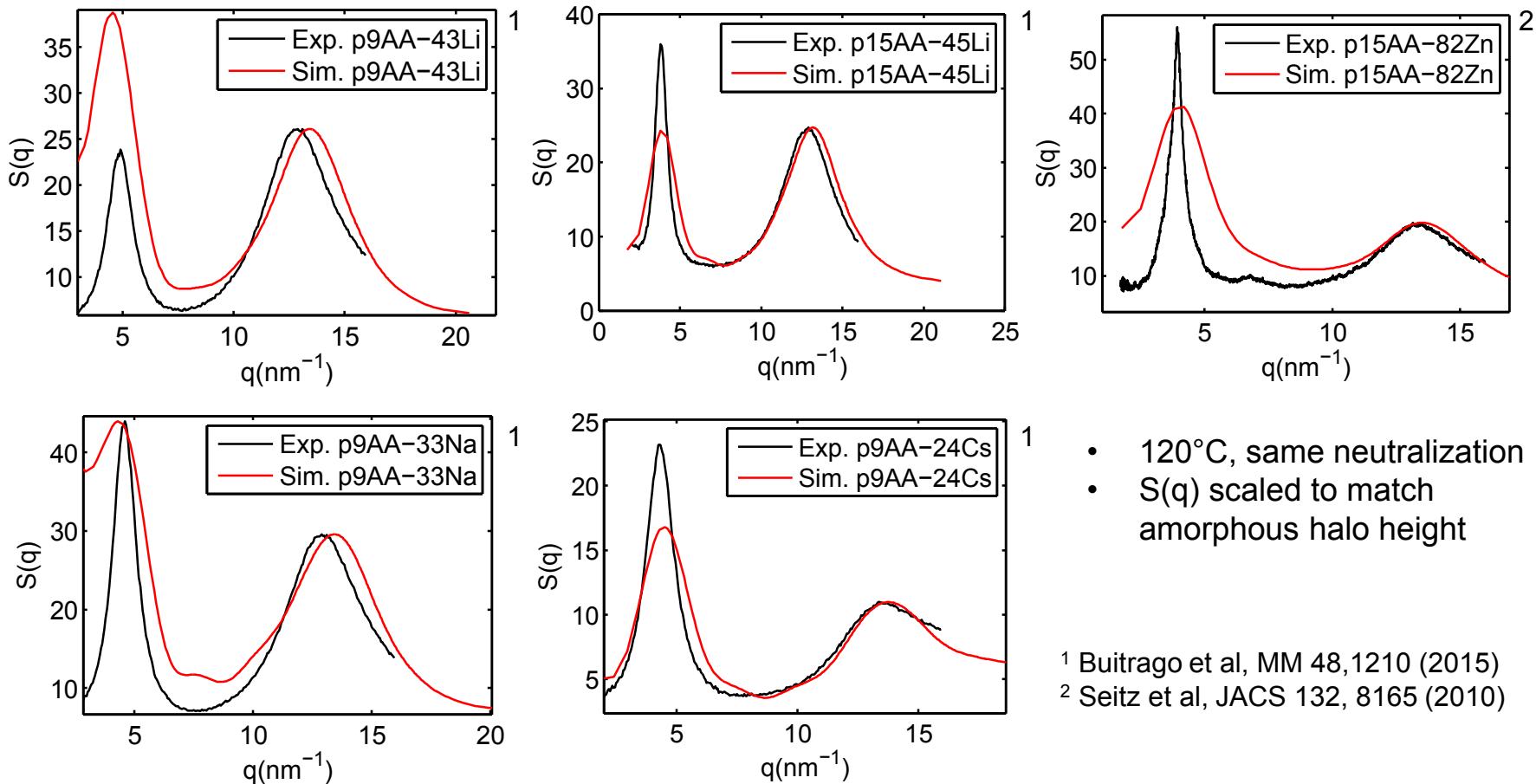
f_i = atomic scattering functions for x-rays

$$f_i = c_i + \sum_{k=1}^5 a_{ik} \exp(-b_{ik} q_0^2)$$

coefficients are tabulated



Direct Comparison to X-ray Scattering



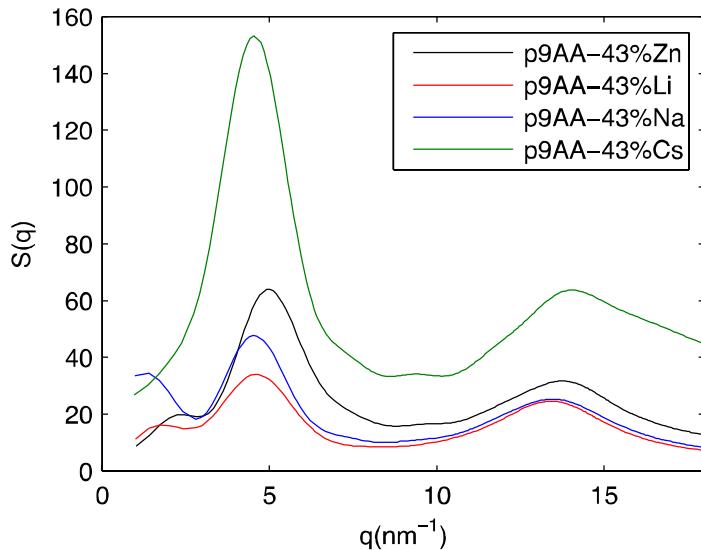
- 120°C, same neutralization
- $S(q)$ scaled to match amorphous halo height

¹ Buitrago et al, MM 48,1210 (2015)

² Seitz et al, JACS 132, 8165 (2010)

- excellent agreement in peak positions
- good agreement in peak shapes

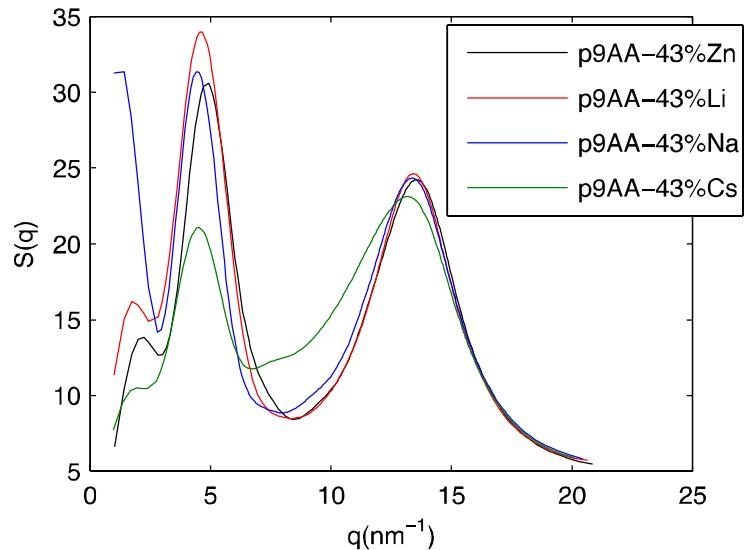
Extra insight from simulation



alchemy
Li⁺ form factor
for all ions



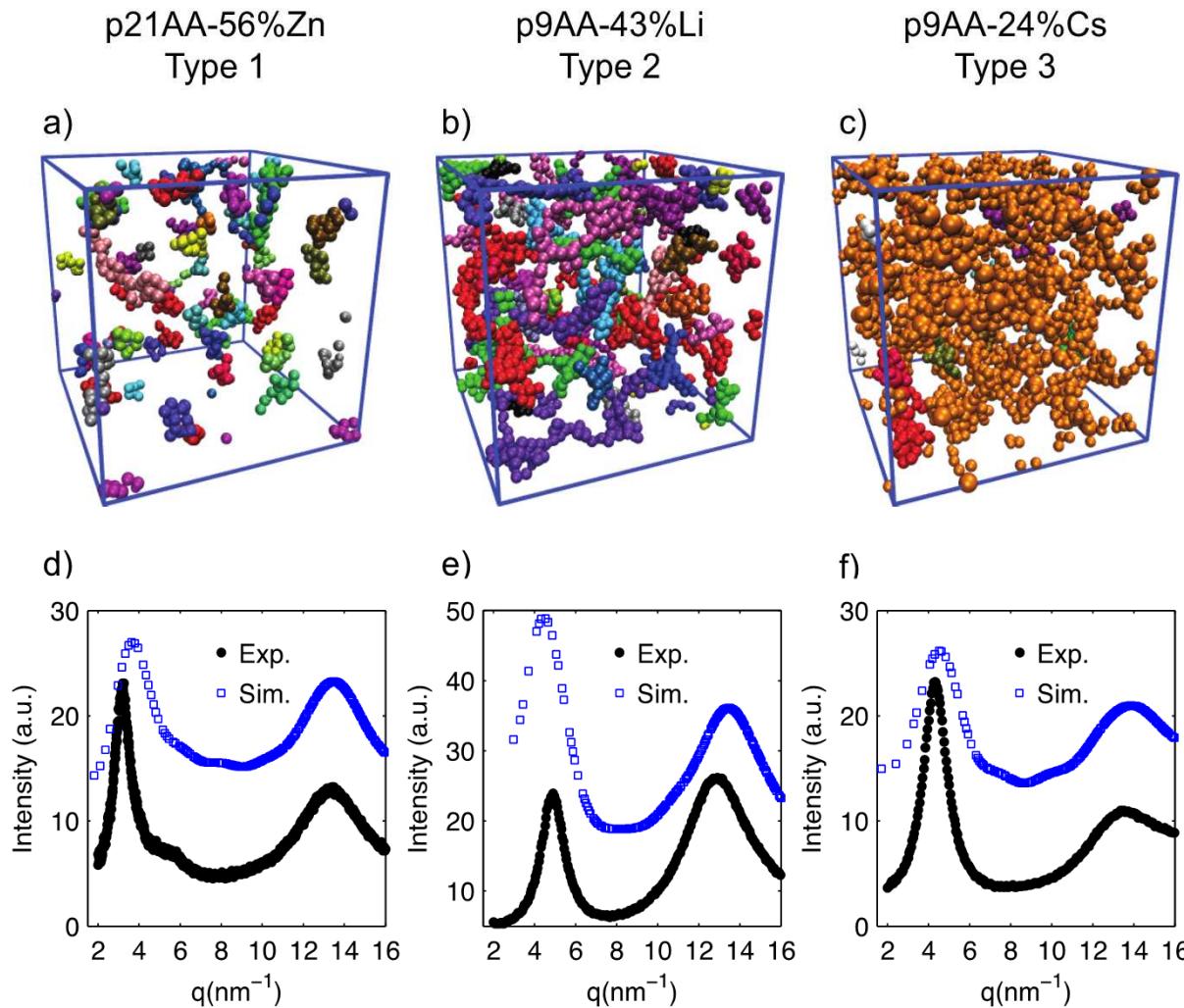
$$f_M \rightarrow f_{Li}$$



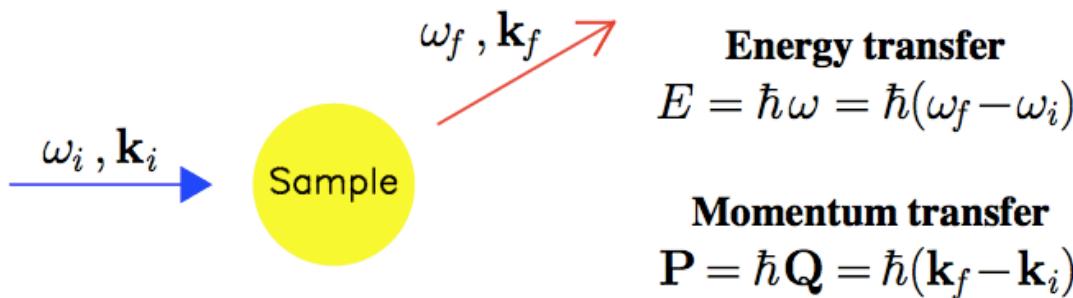
- many differences due to contrast only
- Cs changes short-range structure

X-ray Scattering Doesn't Determine Morphology

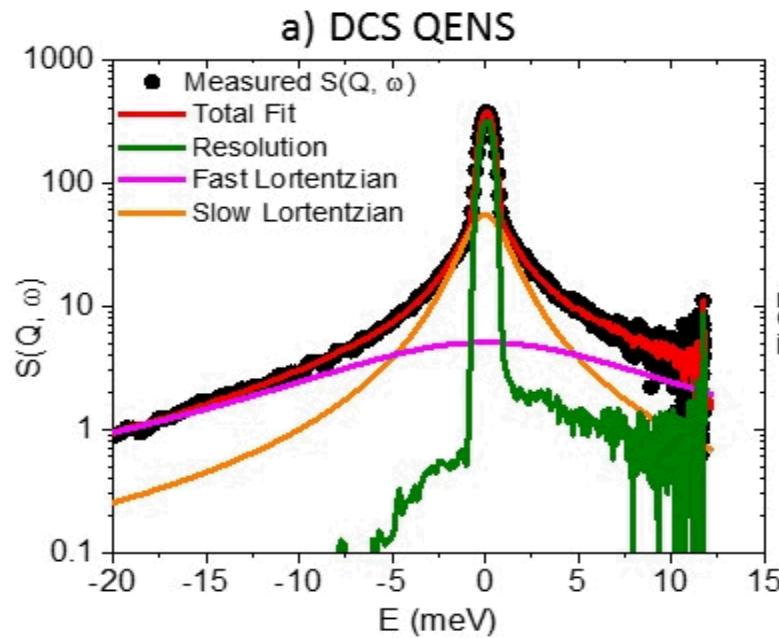
need simulations or imaging!



Quasi-Elastic Neutron Scattering



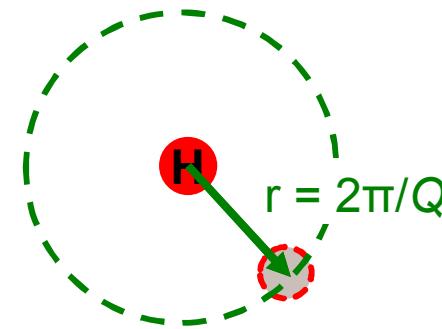
incoherent, inelastic: sensitive to self-motion of hydrogens
scattering intensity proportional to $I(Q, \omega)$



Dynamics: QENS and MD

$$S(Q, t) = \frac{\int S_{exp}(Q, \omega) e^{i\omega t} d\omega}{\int R(Q, \omega) e^{i\omega t} d\omega} \quad S(Q, t) = \int G_s(r, t) \frac{r \sin(Qr)}{Q} dr$$

$G_s(r, t)$: given an atom was at $r=0$ at time $t=0$, the probability that the atom is at r at time t



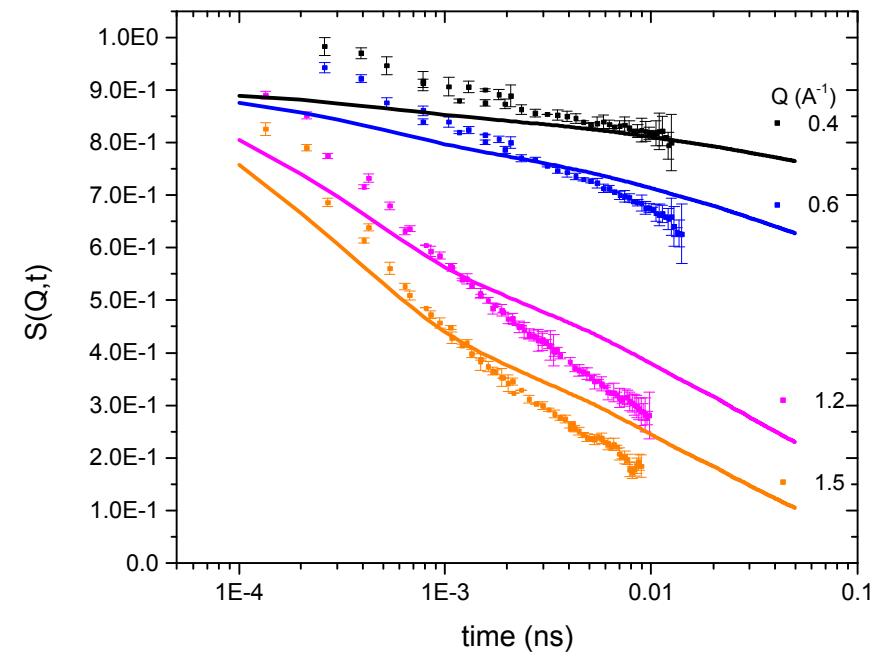
MD simulations

- calculate self-part of van Hove correlation function $G_s(r, t)$ for hydrogens
- exclude chain ends

Comparing to QENS to check force field

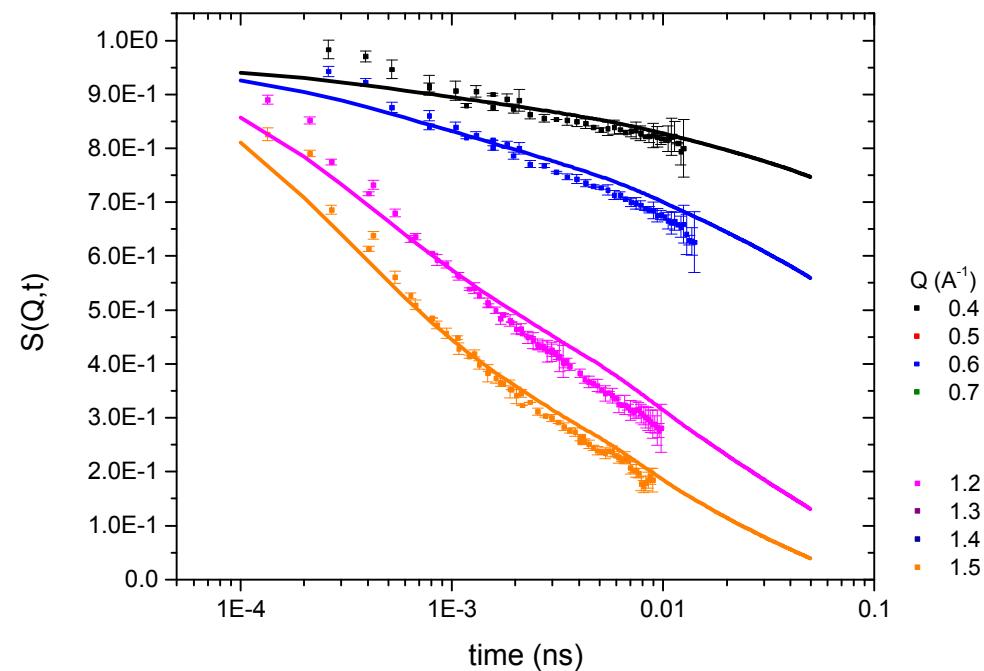
OPLS-AA

p9AA S(Q,t)

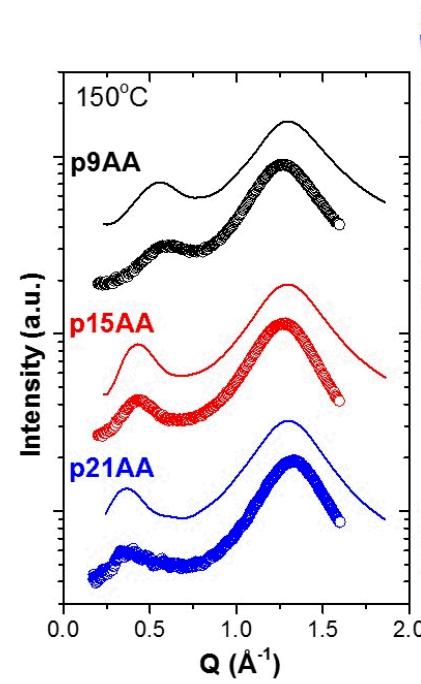
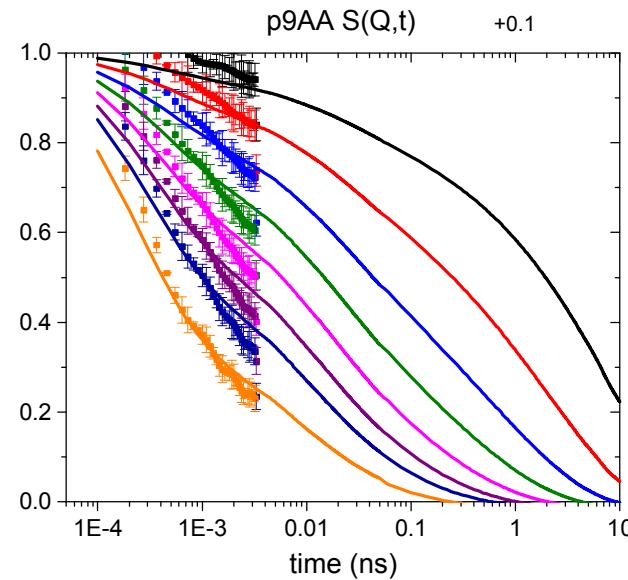
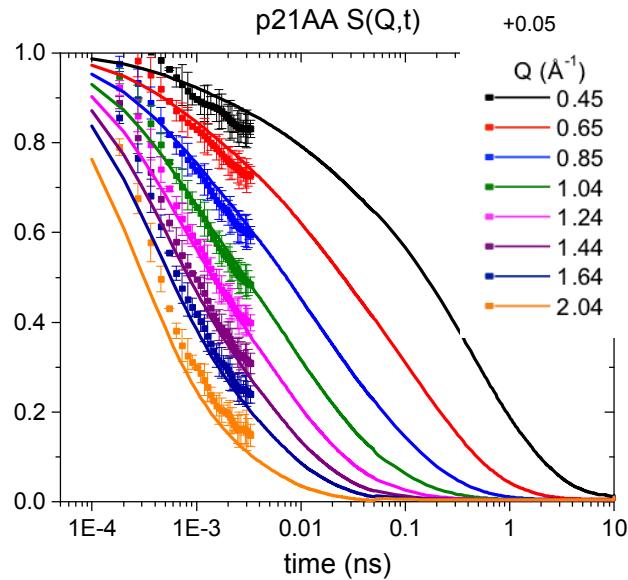


LOPLS-AA

p9AA_Siu S(Q,t)



Acid Copolymers



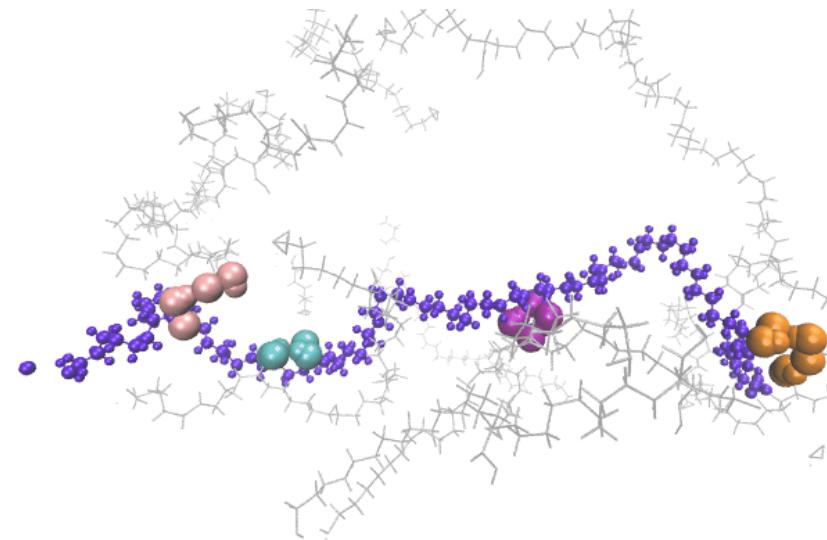
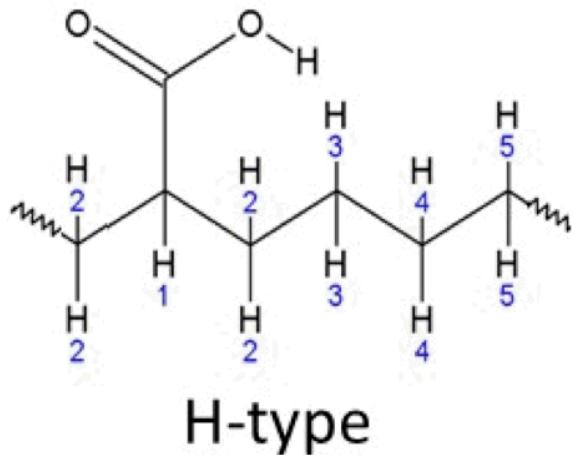
excellent agreement between QENS and MD

relevant length scales:

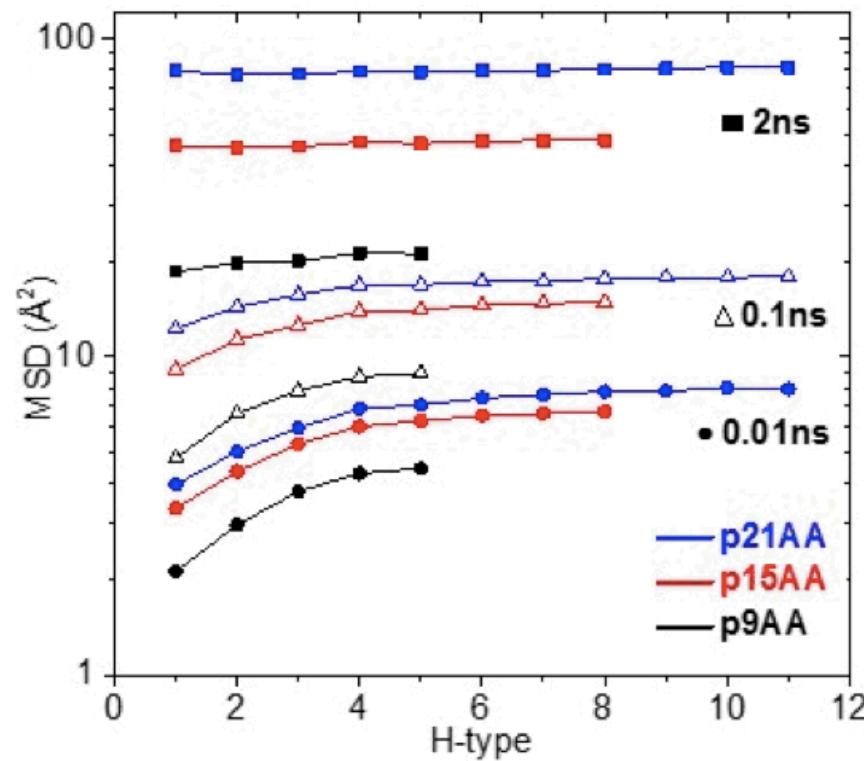
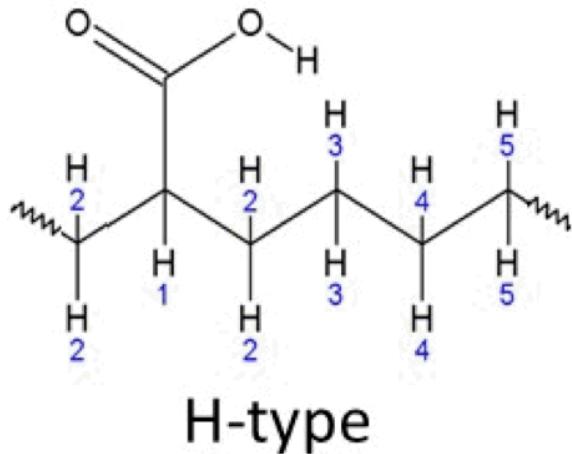
amorphous halo: $Q \approx 1.35 \text{\AA}^{-1}$

ionomer peak: $Q \approx 0.3 - 0.6 \text{\AA}^{-1}$

Local Dynamics



Local Dynamics



short times: H1 motion much slower than middle of chain
 long times: aggregates rearrange, all H motions similar

Outlook

Atomistic MD can answer lots of questions in soft matter

Vast field with constantly evolving methods/codes

For related talks this week:

Focus sessions on “Advances in Molecular Dynamics Simulation: From Atomistic to Coarse-Grained Models”, sponsored by DCP

A25: Monday 8 am

C25: Monday 2:15 pm

F26: Tuesday 11:15 am

K25: Wednesday 8 am

References

Books

- M. P. Allen and D. J. Tildesley, *Computer Simulations of Liquids*, Clarendon Press, Oxford, 1987
- David Frenkel and Berend Smit, *Understanding Molecular Simulation: From Algorithms to Applications*, Academic Press, 2002
- Tamar Schlick, *Molecular Modeling and Simulation: An Interdisciplinary Guide*, Springer, 2002

Others

- Frenkel, D. (2013). Simulations: The dark side. *The European Physical Journal Plus*, 128(1), 10
- https://en.wikipedia.org/wiki/Comparison_of_software_for_molecular_mechanics_modeling
- Arbe, A., Alvarez, F., & Colmenero, J. (2012). Neutron scattering and molecular dynamics simulations: synergetic tools to unravel structure and dynamics in polymers. *Soft Matter*, 8(32), 8257