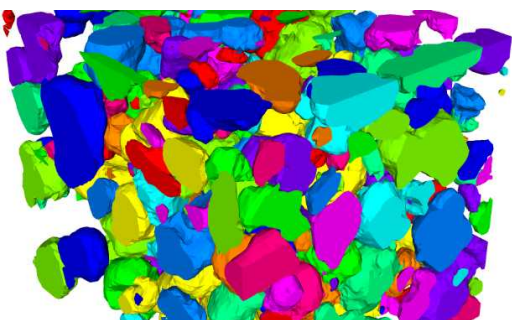
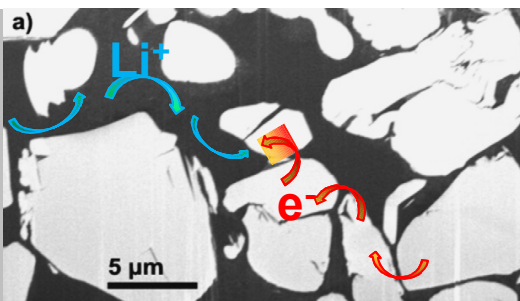


Conductivity Degradation of PVDF Binder During Cycling: Measurements and Simulations for Lithium Ion Batteries



Anne M. Grillet, Thomas Humplik, Emily K. Stirrup,
David A. Barringer, Scott A. Roberts, Chelsea M.
Snyder, Bradley Trembacki & Chris A. Apblett

AIChE Annual Meeting 312e
November 15, 2016

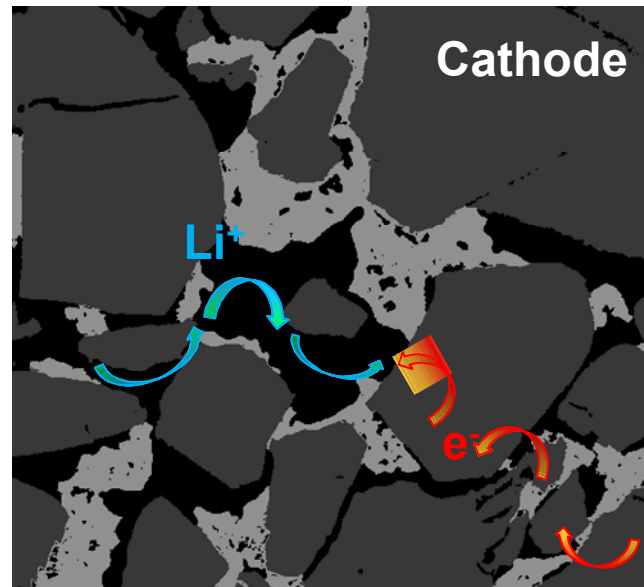


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SAND2016-4743 C

Battery Introduction

- Battery electrodes are multicomponent particle composites
 - Binder is a mixture of polyvinylidene fluoride (PVDF) and carbon black
 - Binder is important for mechanical stabilization and electrical conduction
 - Electronic Conductivity : Binder \sim graphite \gg LiCoO_2

SEM image of LiCoO_2 cathode



94wt% LiCoO_2
 3wt% PVDF
 3wt% CB } "Binder"
 electrolyte

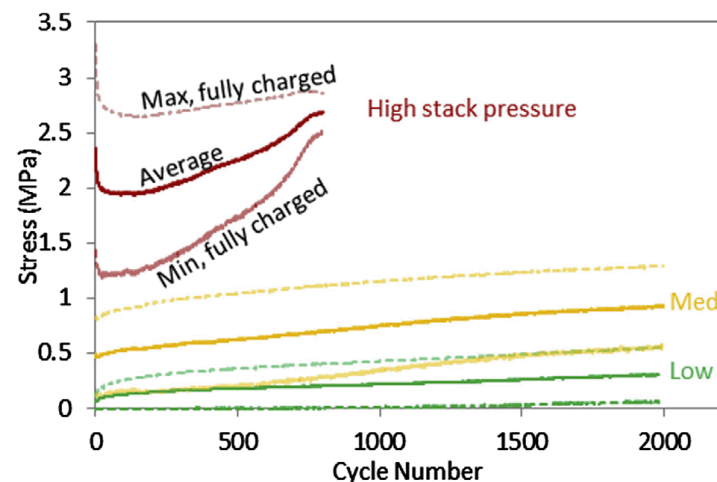
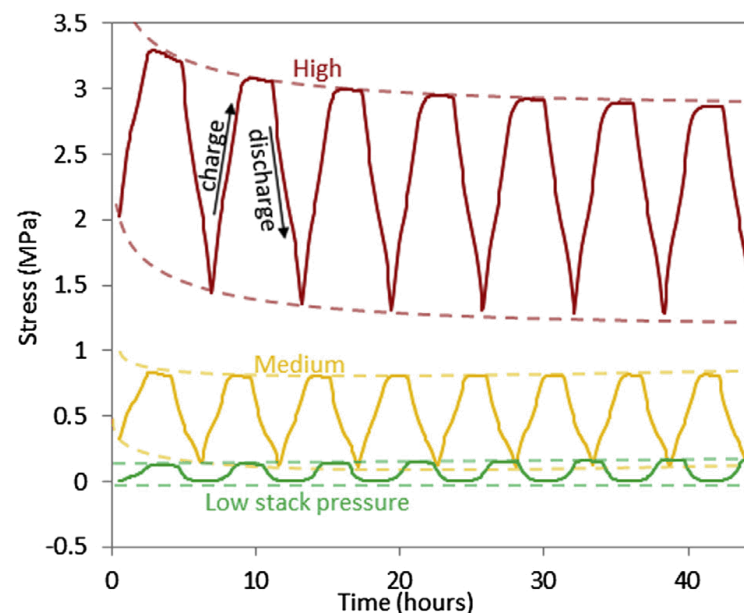
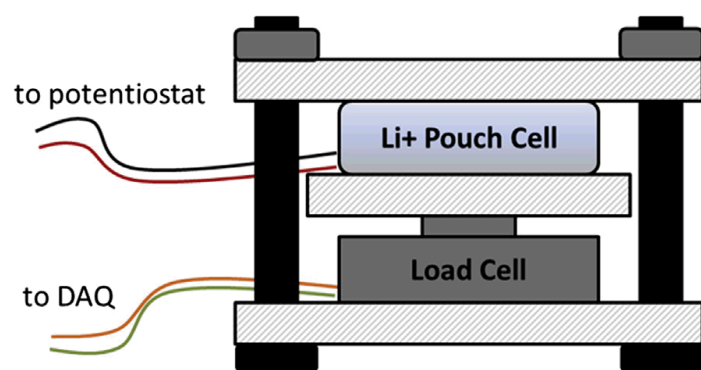
Battery voltage depends on how efficiently lithium ions and electrons are transported

Outline

- Battery electrode binders play an important role both mechanically & electrochemically
 - Coupled mechano-electrochemical characterization of binder
 - Mechanical stresses to mimic stresses during electrochemical cycling
 - Measure electrode strain and electrical conductivity
 - PVDF/CB binder films
 - LiCoO₂ cathodes
 - Comparison to electrochemically cycled LiCoO₂:Graphite batteries
 - Mesoscale simulations of composite cathodes with binder
- Mechanical cycling of binder causes increased internal resistance which degrades battery performance

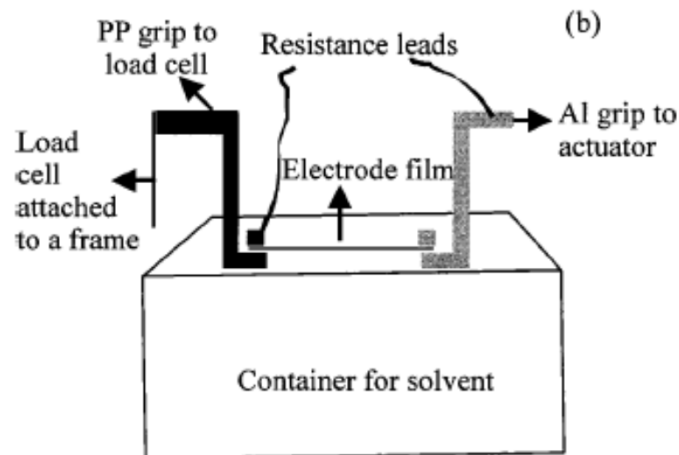
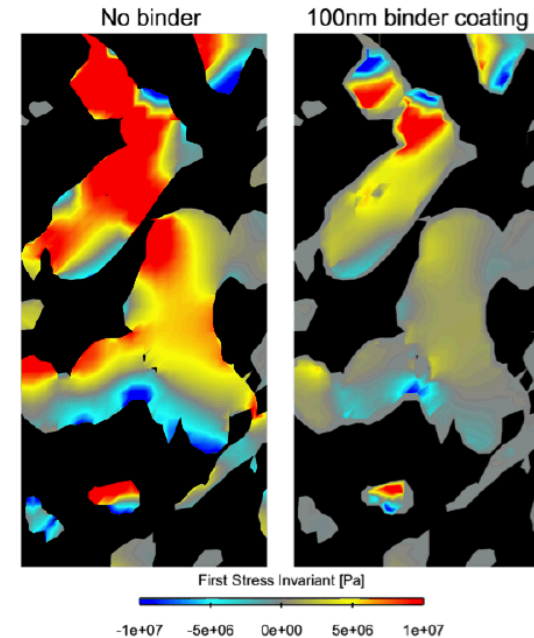
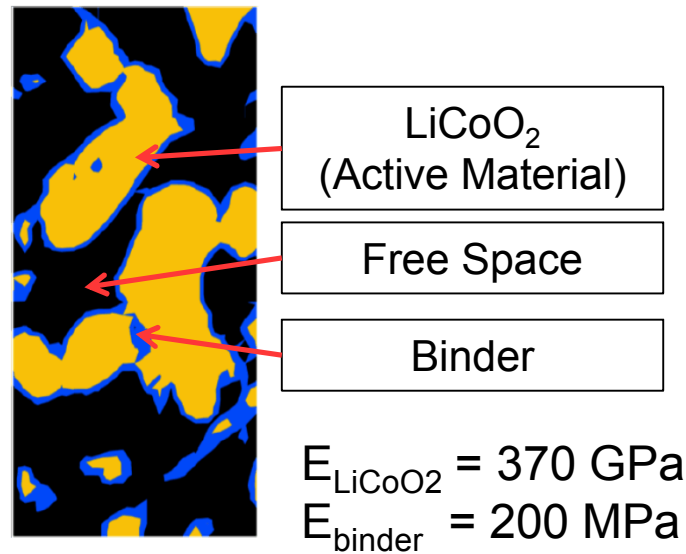
Both Electrodes Swell during Charging

- Constrained battery pouch cell experiments by Cannarella & Arnold
 - See large changes in stress during cycling as cathode and anode swell during charge and shrink during discharge
 - Electrodes want to swell after repeated cycling resulting in increased stresses with time



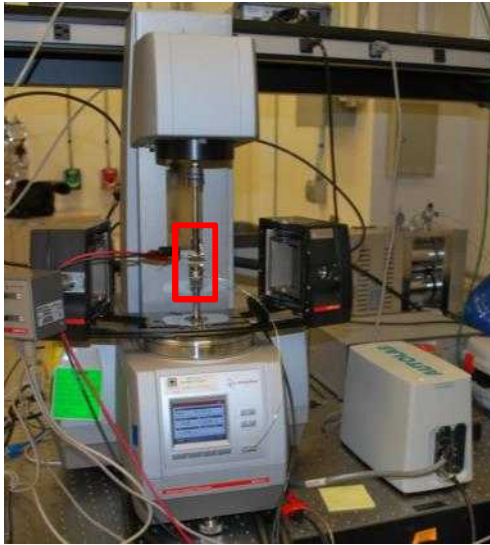
Role of Binder in Battery Cathodes

- Mendoza et al.¹ used experimental cathode microstructures to predict peak stresses during charging of 100's MPa
- Presence of 100nm binder could reduce stresses by 50%

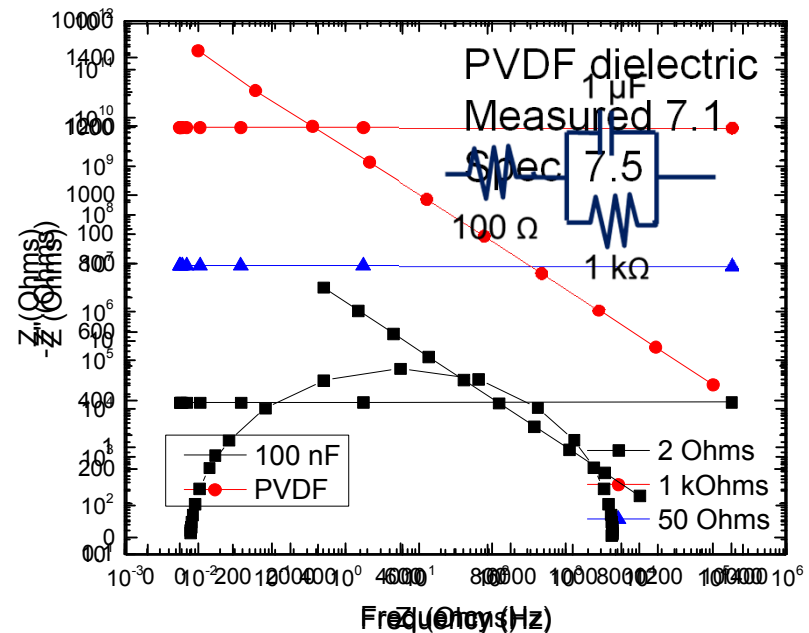


- Chen *et al.*² investigated the change in resistance of various PVDF/carbon composites (both wet and dry) and during initial cycling in tension
 - Small changes in resistivity with ~5 cycles number

Mechanics/Impedance Measuring System

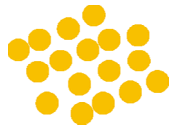


- Measure electrical properties of system during mechanical tests
 - Anton Paar MCR 502 rheometer
 - Compression of binder disks
 - Metrohm PGASTAT204/FRA 32
 - Potentiostat/galvanostat 10 μ Hz - 1 MHz
 - Welded wire leads to plates to reduce lead resistance
 - Performing Electrical Impedance Spectroscopy
- Qualified that combined tool can measure a variety of electrochemical systems

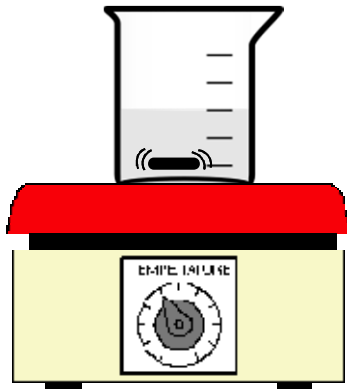


Synthesis of PVDF/CB Binder films

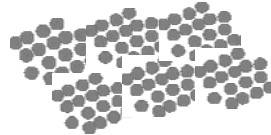
Solvay 5130 PVDF



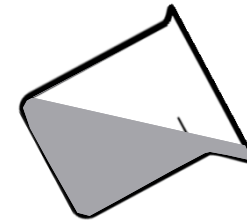
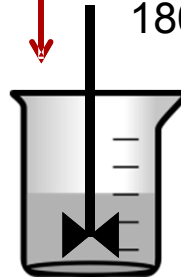
Dissolve in
N-Methyl-Pyrrolidone
 $T=50-70^{\circ}\text{C}$



Denka Carbon Black



Mix into solution
1800RPM 2hrs

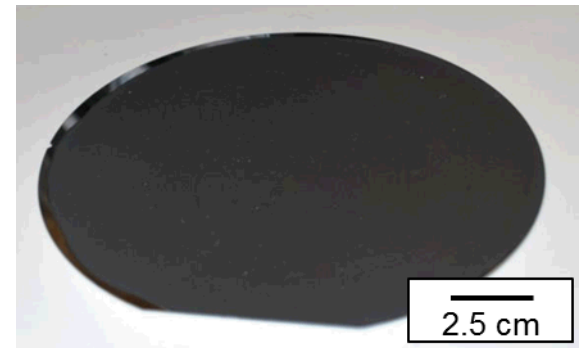


Pour onto
Si wafer

Dry in vacuum
oven at 110°C for
12 hours



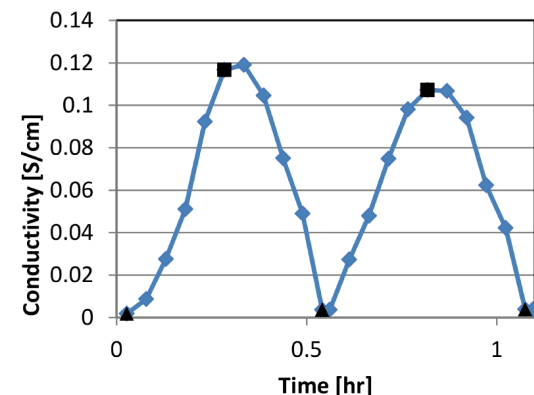
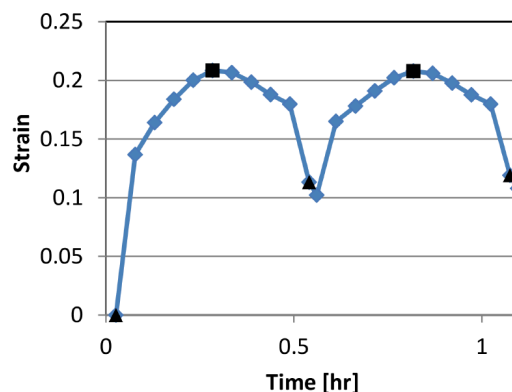
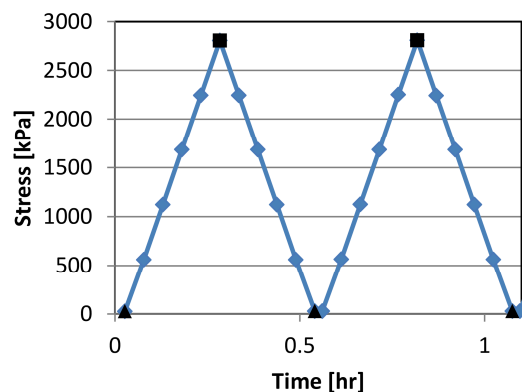
- Film thickness on the order of $70 - 200\ \mu\text{m}$



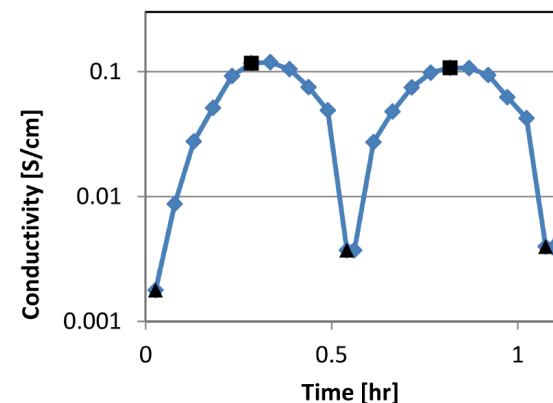
Mechanical Cycling of Dry Binder Film

PVDF/CB composite (30wt% CB)
(6x 5 mm diameter discs, ~500 μm thick)

Procedure:
28 kPa – 2.8 MPa – 28 kPa at
~4C cycle rate

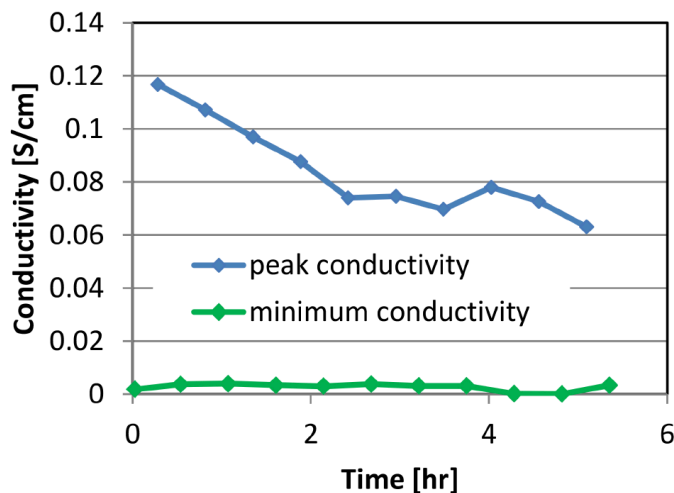
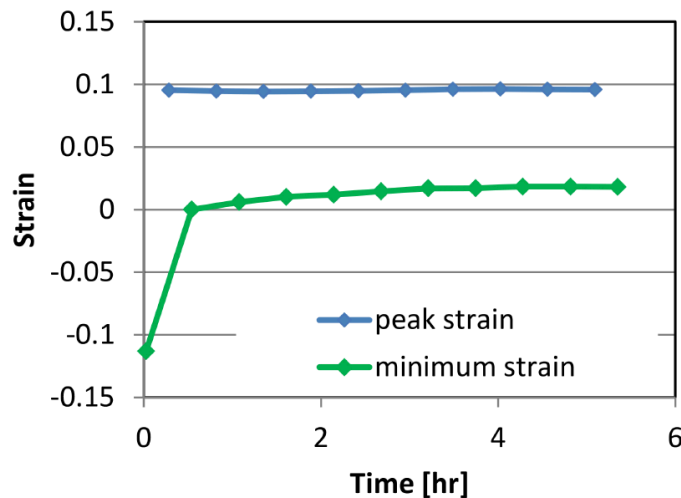


- Plastic deformation after first cycle (flattening of disks or surface non-uniformities)
- Electrical conductivity exhibits a strong dependence on applied load (and resulting strain)
 - Changes by two orders of magnitude



Mechanical Cycling of Dry Binder Film

10 cycles at 4C

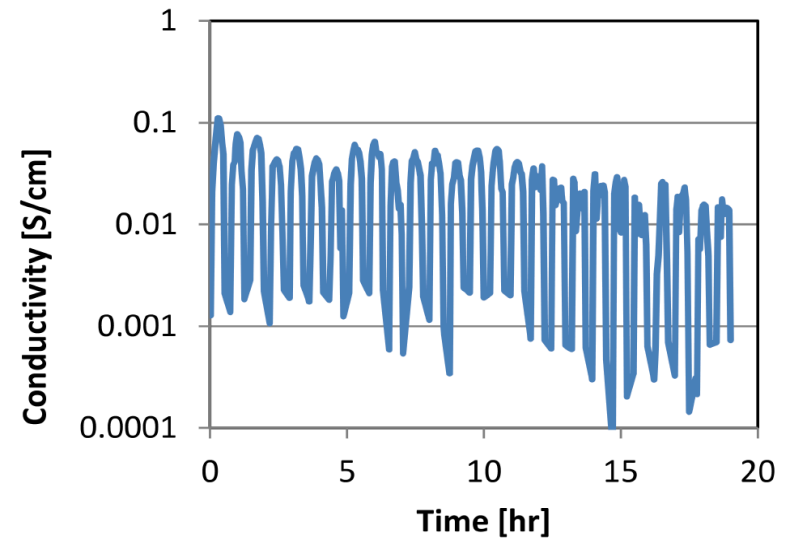
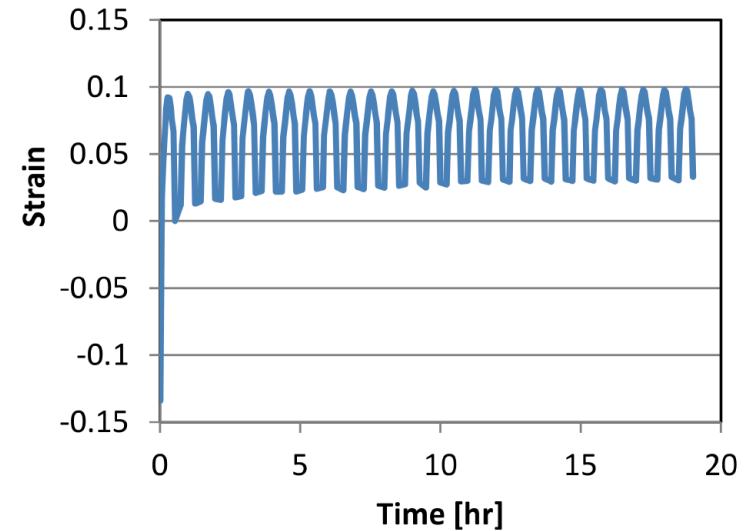
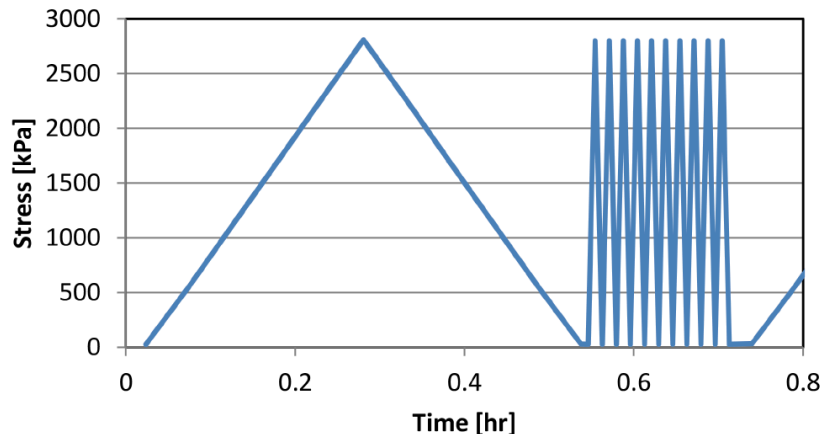


- Mechanical cycling
 - 27 kPa – 2.8 MPa – 27 kPa
 - 4C (cycle \approx 32 min)
 - 10 cycles (5+ hours)
- Slow consolidation of binder film
- Significant 30% decrease in binder electrical conductivity after 10 cycles
- SEM imaging of binder films before and after cycling show no obvious morphology changes (cracks, delamination from the particles, etc.)

Mechanical Cycling of Dry Binder Film

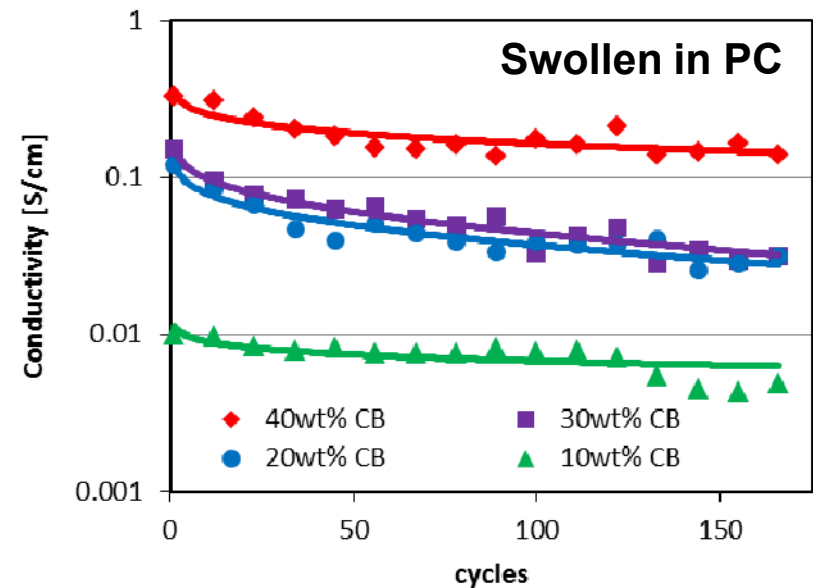
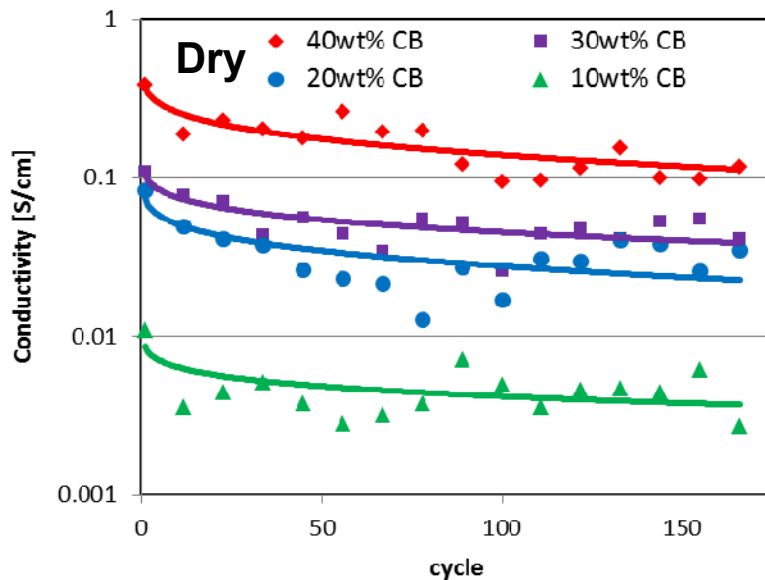
Accelerated cycling

- 10 cycles at 120C between each measurement cycle at 4C
 - 28 kPa – 2.8 MPa – 28 kPa
 - 4C (cycle \approx 32 min), 10x 120C (cycle \approx 1min)
 - 266 total cycles (19 hours)
- 30wt% Carbon black dry binder film



Mechanical Cycling of Swollen Binder Film

- Examined affect of carbon black concentration
- Binder absorbs 20-40wt% of propylene carbonate (PC) solvent
- Cycled using same accelerated cycling protocol
 - 4C (cycle \approx 32 min), 10x 120C (cycle \approx 1 min)

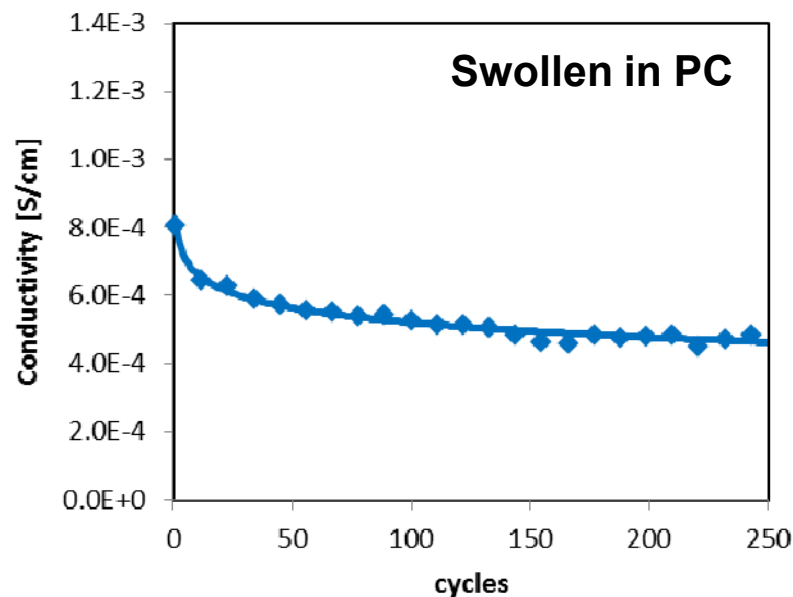
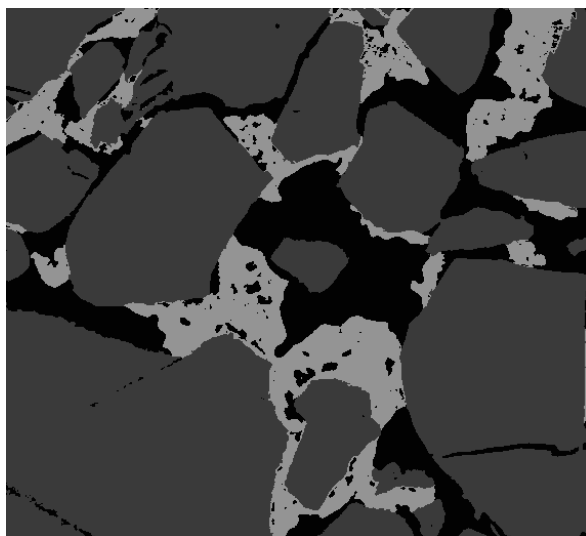


- Significant decrease in binder conductivity as a function of mechanical cycling between 45-75%
- Same for both Solvay 5130 and Kureha W #1300 battery grade PVDF polymers

Mechanical Cycling of LiCoO_2 Cathode

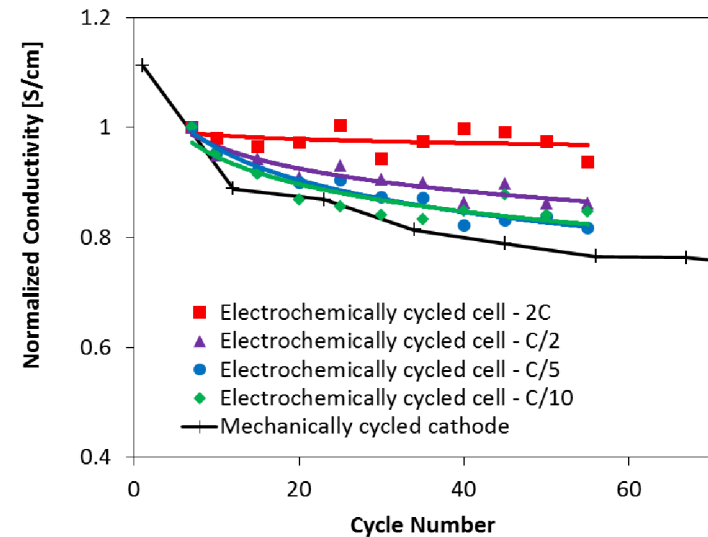
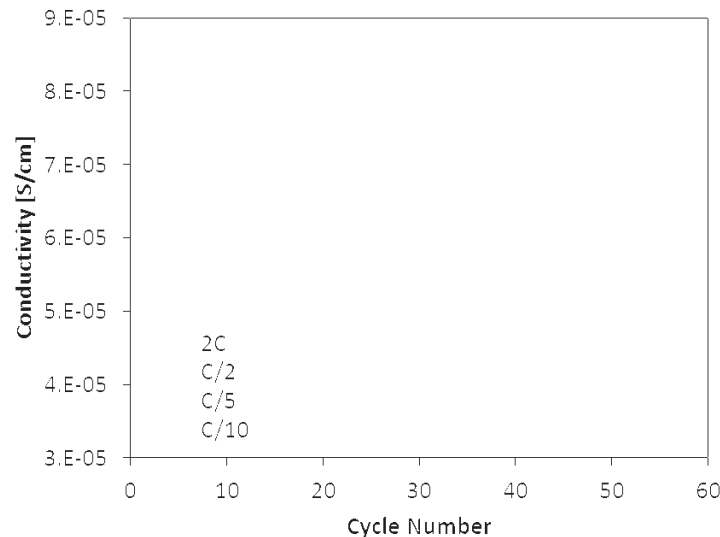
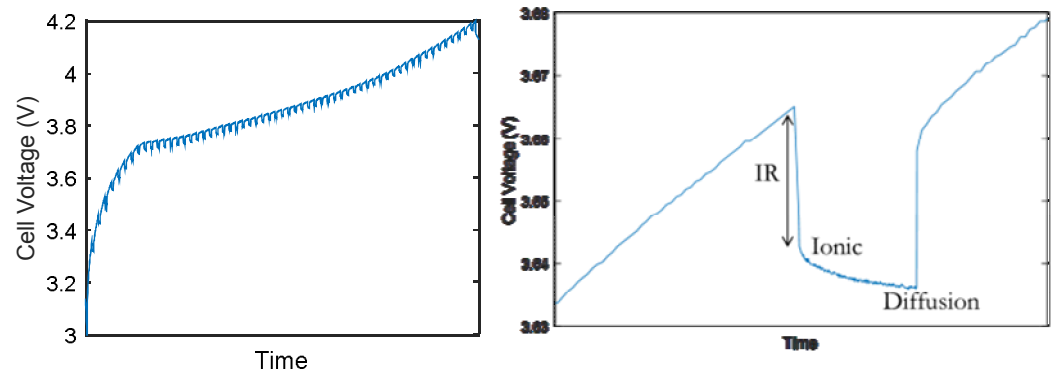
- Cathode depends on the binder for reducing internal resistance
 - Fresh LiCoO_2 active material: $\sigma_{\text{LiCoO}_2} = 10^{-6} \text{ S/cm}$ vs 0.5-0.05 S/cm for binder
 - Cycled LiCoO_2 active material: $\sigma_{\text{LiCoO}_2} = 10^{-3} \text{ S/cm}$
 - Propylene carbonate solvent: $\sigma_{\text{PC}} = 10^{-10} \text{ S/cm}$
- Similar trend in degradation of electronic conductivity of cathode: peak electronic conductivity declined by 42% after 166 mechanical cycles

94wt% LiCoO_2
6wt% PVDF/CB
solvent



Electrochemical vs Mechanical Cycles

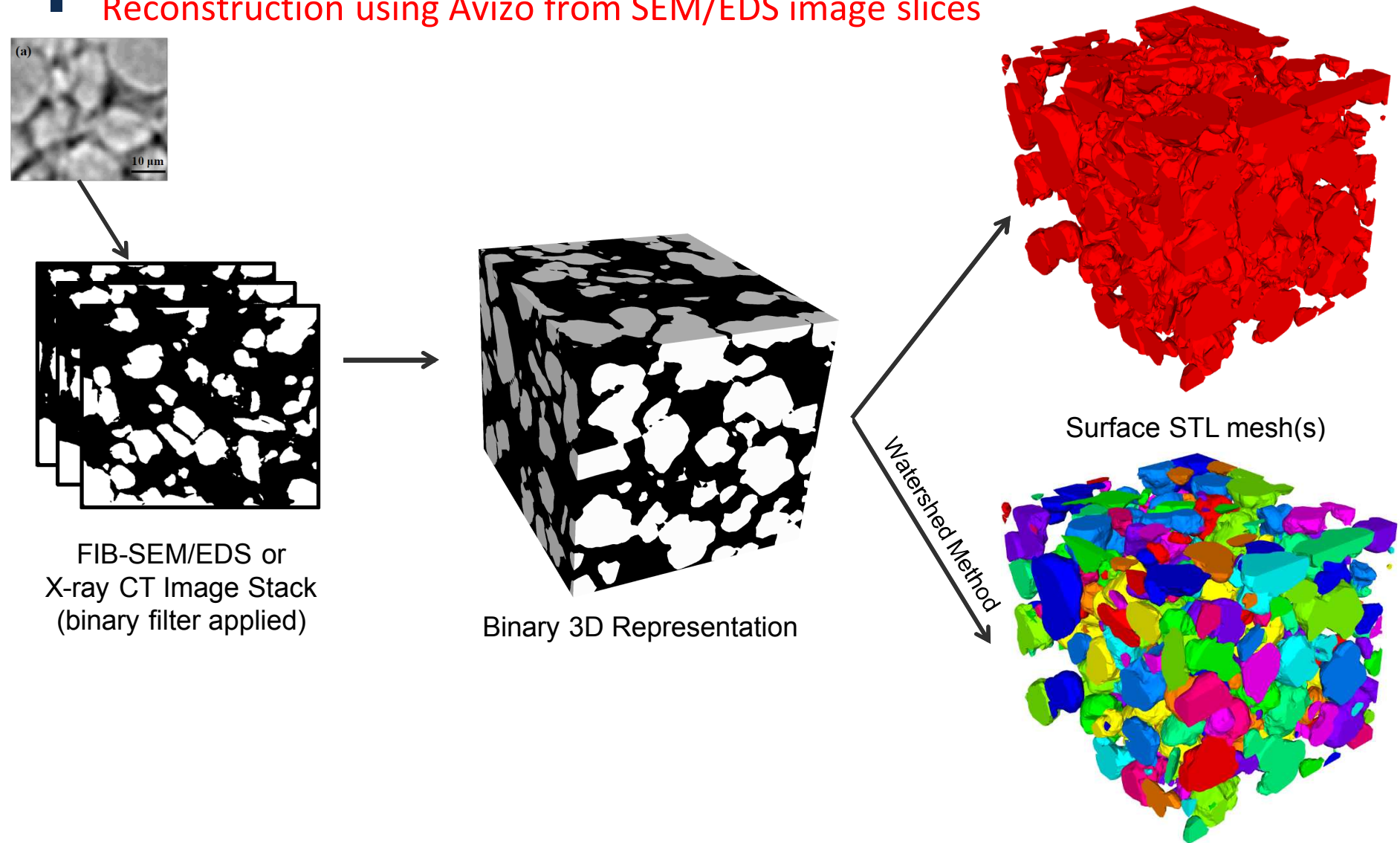
- Current interrupt testing to measure ohmic resistance of full cell between 3-4.2 V
- Ohmic resistance starts as 61% of total cell resistance at C/10 charging rate



- Between cycle 7-57 cycles, electrochemical cycling reduced cell conductivity by 12-26% compared to a 31% decrease for mechanical cycling of the cathode

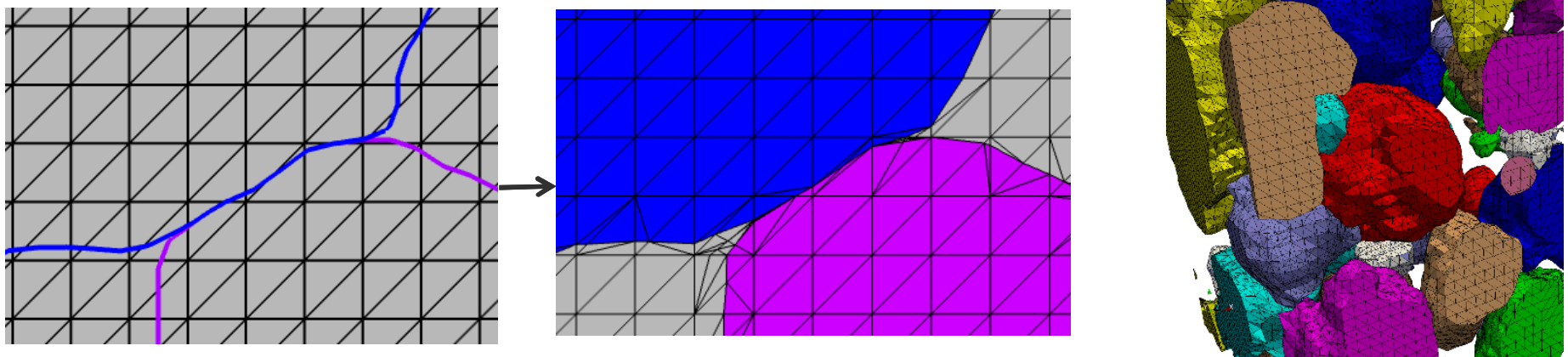
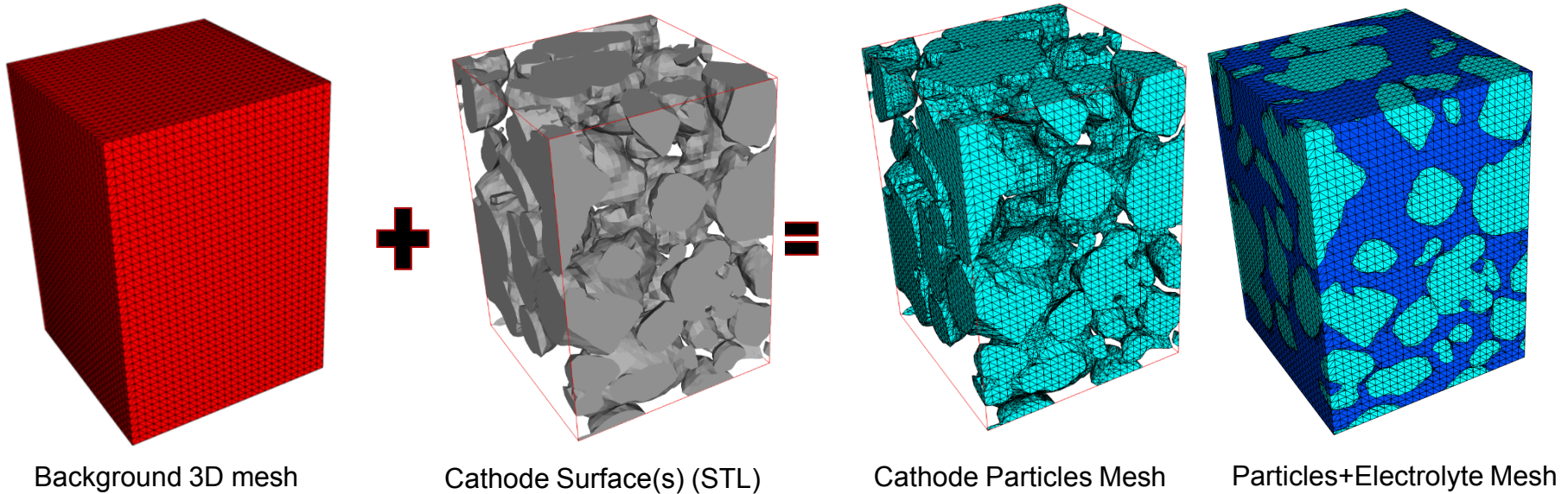
Reconstruction Approach

- Reconstruction using Avizo from SEM/EDS image slices



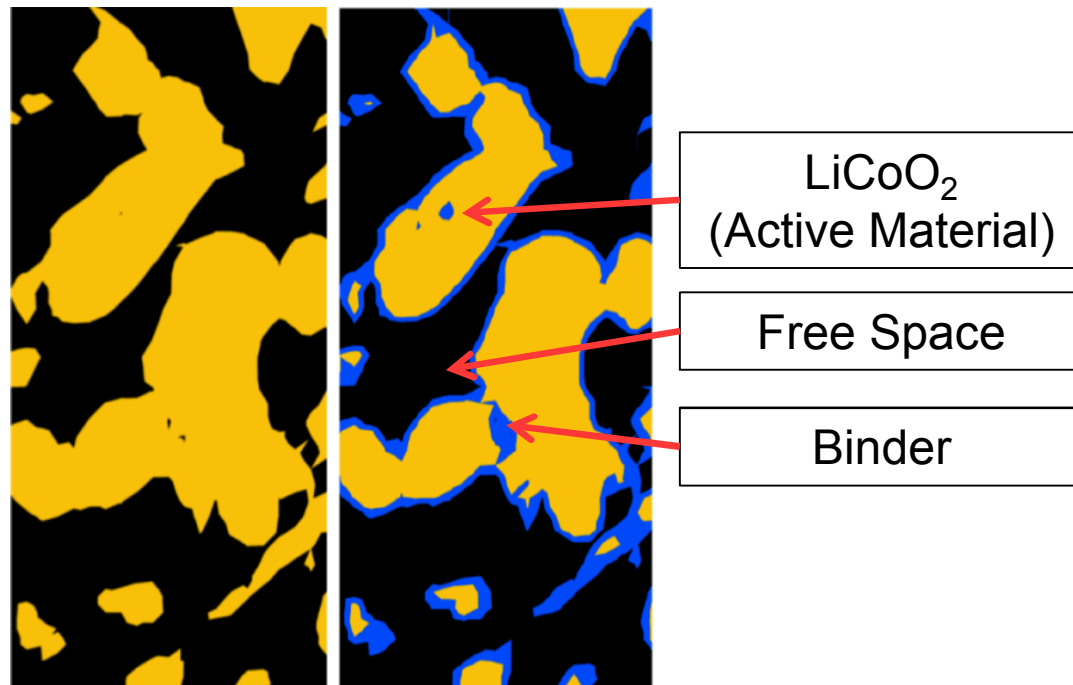
Reconstruction Approach

- Conformal Decomposition FEM (CDFEM), (Sandia Method: Noble 2010)

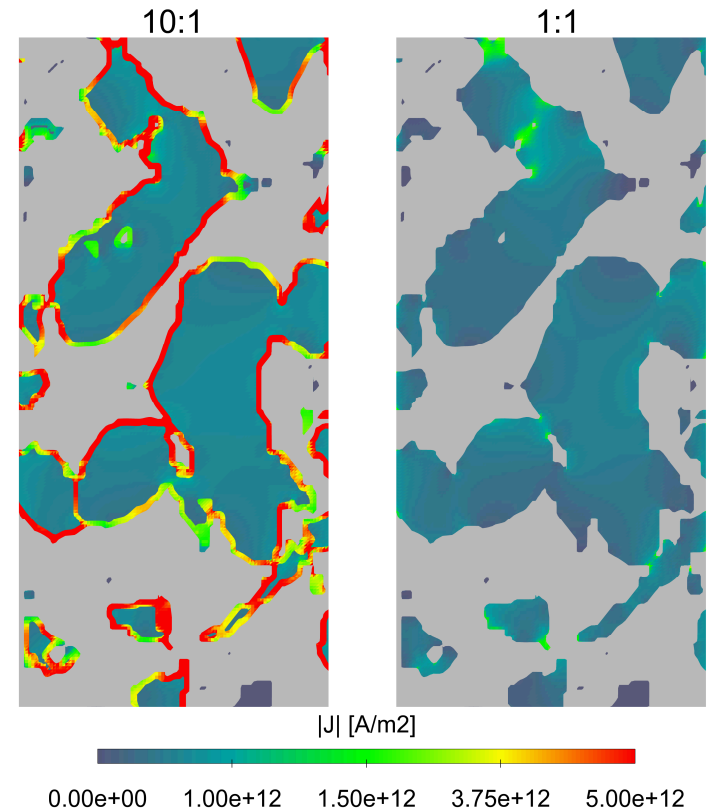


Electrical Conductivity in a Cathode

- Calculate the effective electrical conductivity of a representative cathode microstructure as a function of the binder conductivity.
- Highest current densities are always located in the binder phase.

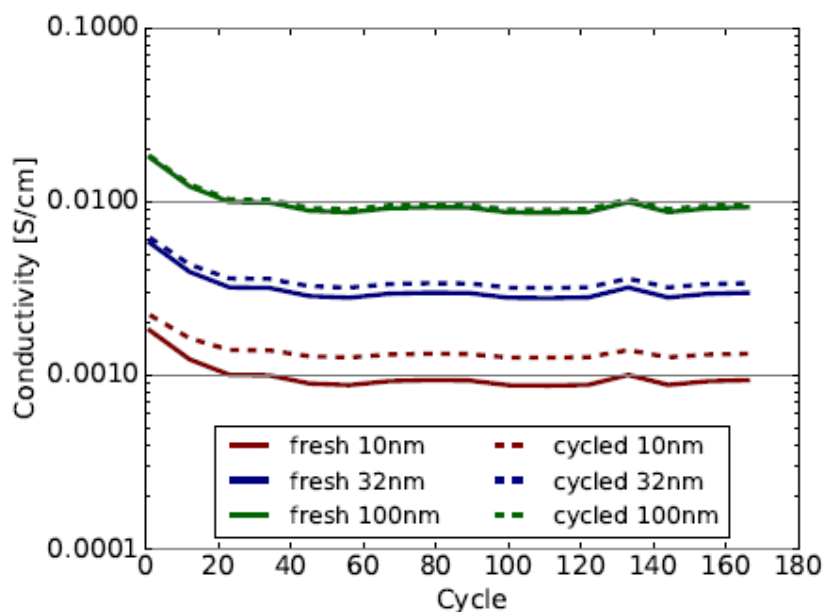
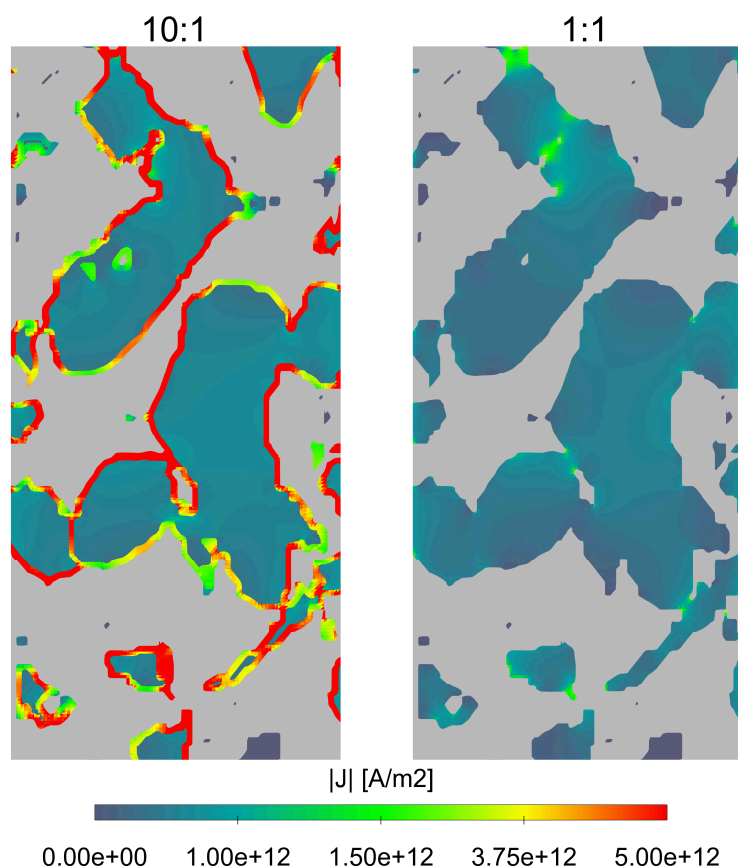


Binder:LiCoO₂ conductivity



Impact of Binder Conductivity Decline

- Calculate the effective electrical conductivity as the binder conductivity decreases.
- Based on cathode composition, 32nm coating of binder on active material
 - For 40wt% CB, predict 56% decline in cathode peak electrical conductivity compared to 42% decrease in the experiment

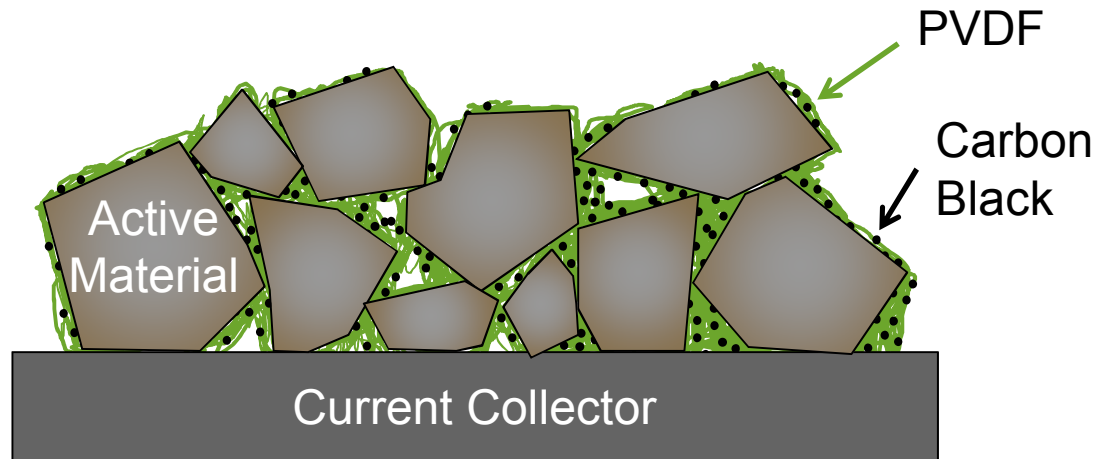


- Binder conductivity changes alone can account for decrease in cathode electrical conductivity

Conclusions

- Mechanical stress cycling causes PVDF/carbon black binder electrical conductivity to degrade 45-75% after 166 cycles
 - Carbon black fractions from 10-40wt%, Dry or swollen with propylene carbonate, Two PVDF manufacturers
- Mechanical stress cycling causes LiCoO_2 cathodes electrical conductivity to decrease by 29-42% after 166 cycles
- Electrochemical cycling causes full cell electrical conductivity to decrease by 12-26% after 50 cycles
 - Cathode conductivity decreased by 31% with mechanical stress cycles
- Mesoscale mechanical modeling shows binder degradation can account for the decline in cathode conductivity
- Mechanical stresses generated during electrochemical cycling degrade the binder electrical conductivity and increase the battery's internal resistance
- Grillet et al. J. Electrochem. Soc. v163(9) (2016) doi: 10.1149/2.0341609jes.
- Roberts et al. Journal of Electrochemical Energy Conversion and Storage v13 (2016) doi: 10.1115/1.4034410.
- Mendoza et al. Electrochem Acta v190 (2016) doi:10.1016/j.electacta.2015.12.224.

Understanding the Role of the Binder



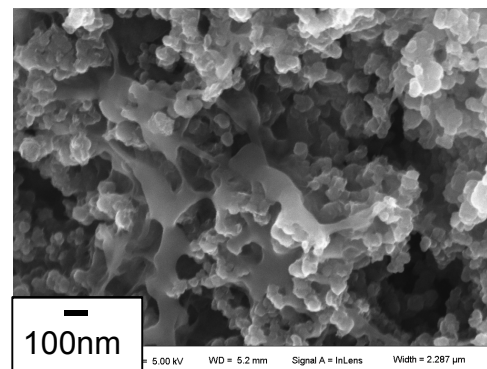
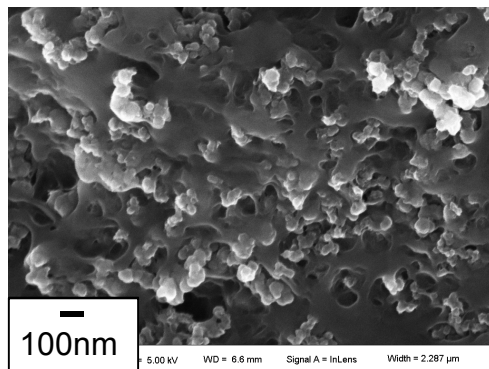
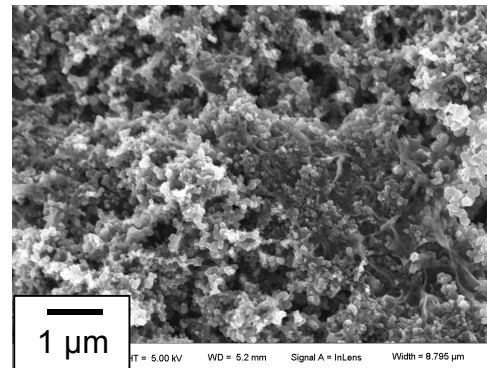
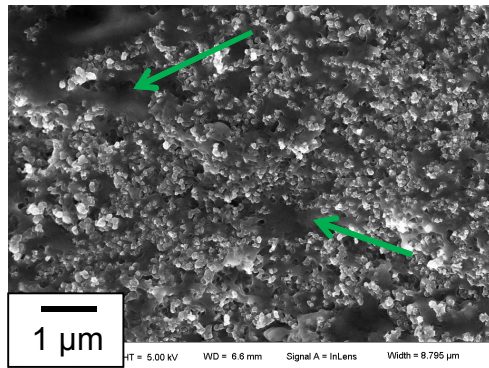
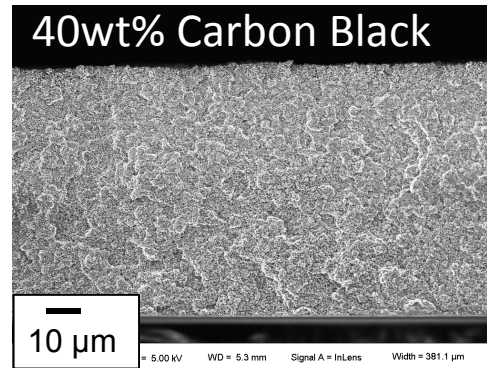
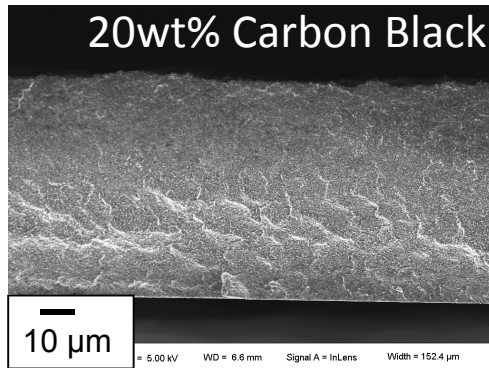
Mechanical

- Mitigating stresses of swelling/contracting active materials
- Maintaining adhesion of active materials to conductive network

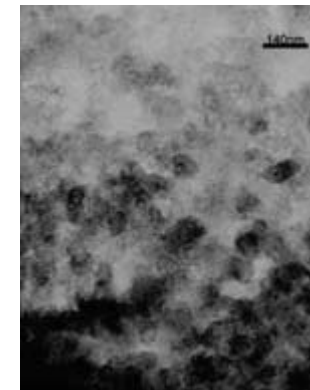
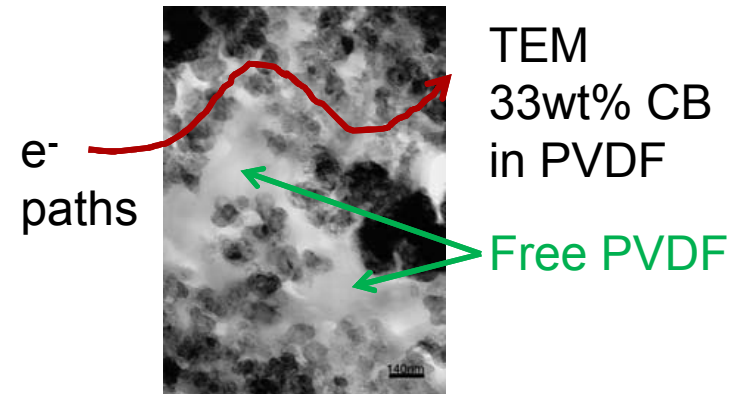
Electrical

- Provide pathway for electron transfer through electrodes
- Decrease resistance (*i.e.*, loss) for cathode

Morphology of PVDF/Carbon Black Composites



- Electron Microscopy of PVDF/Carbon Black Composites – 40% porous
- Lower carbon black fractions have regions of **free PVDF**

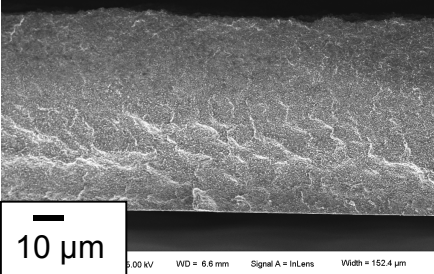


TEM
45wt% CB in
PVDF

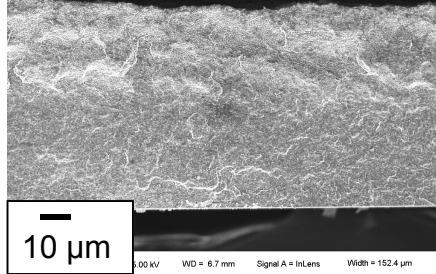
No free PVDF

Morphology of PVDF/Carbon Black Composites

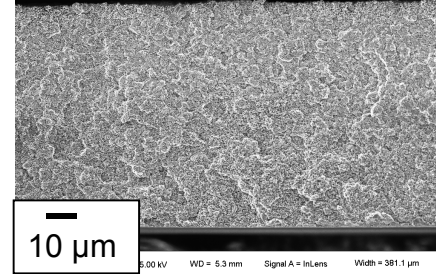
20wt% CB uncycled



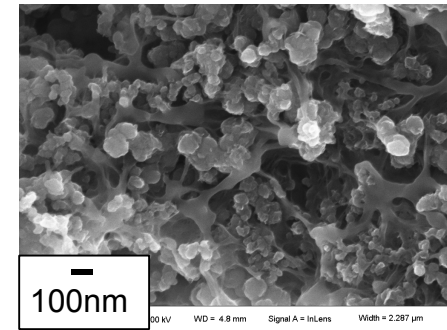
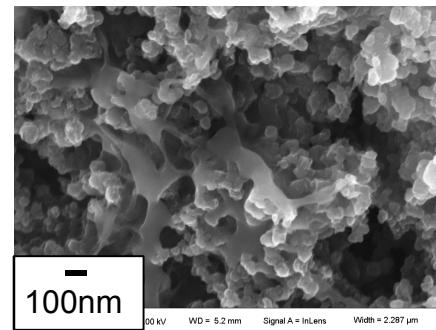
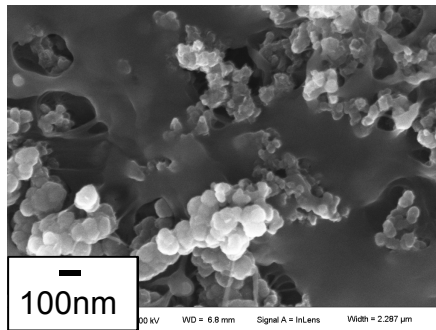
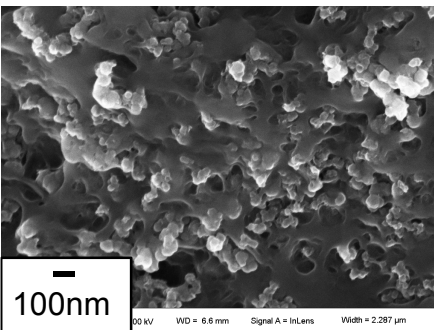
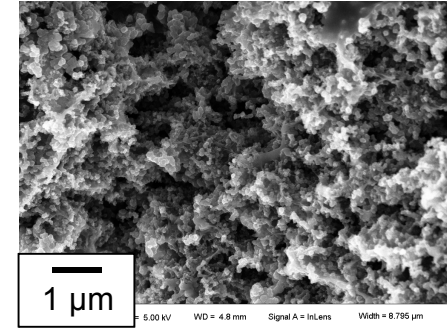
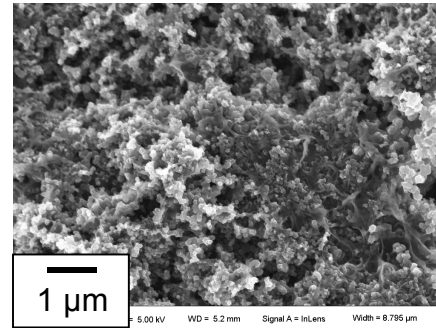
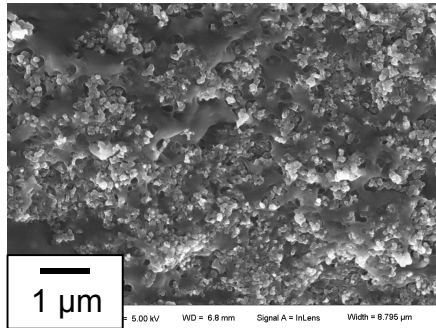
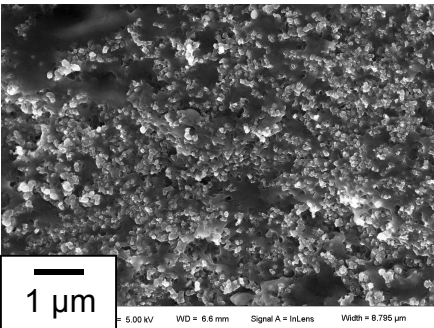
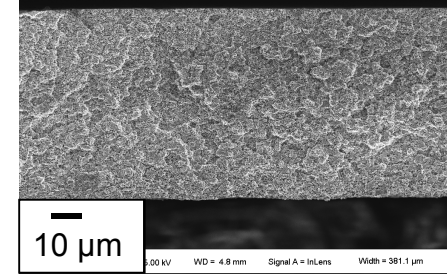
20wt% CB 266 cycles



40wt% CB uncycled



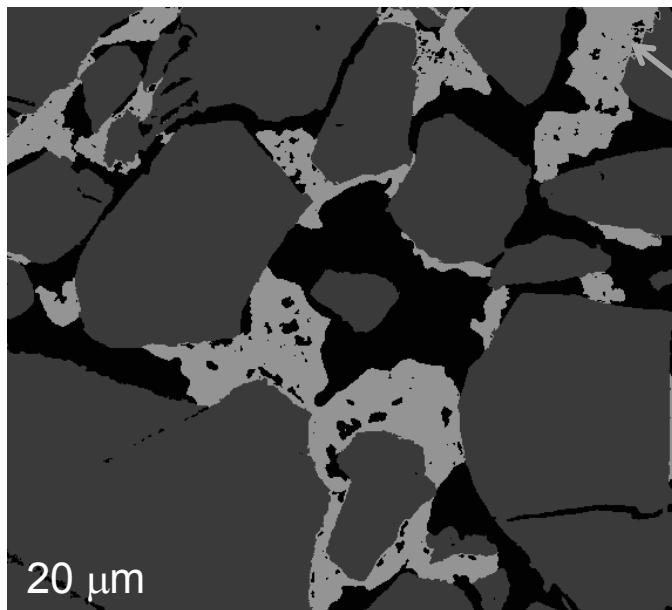
40wt% CB 166 cycles



- No obvious cracks, delamination or other morphology changes

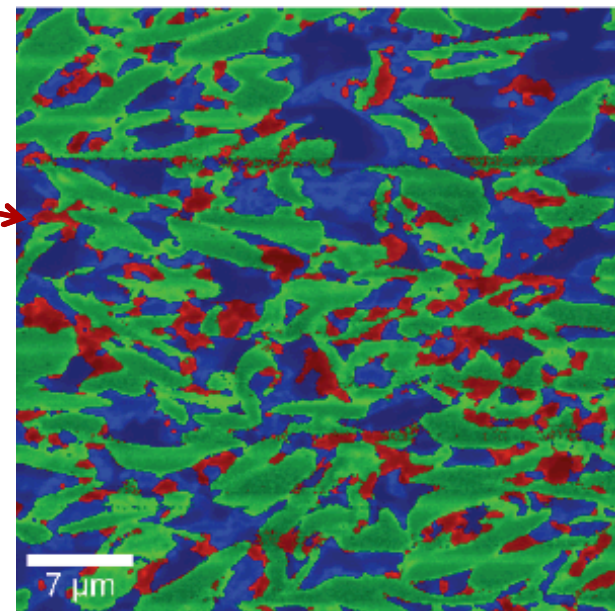
Batteries are actually complex composites

- Electrode active material particles are held together by binder
 - Binder is a mixture of polyvinylidene fluoride (PVDF) and carbon black
 - Electronic Conductivity : Binder \sim graphite \gg LiCoO₂



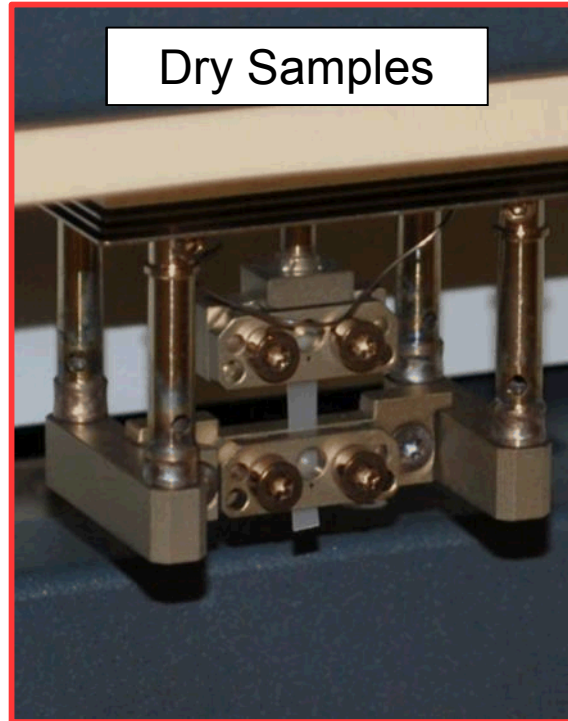
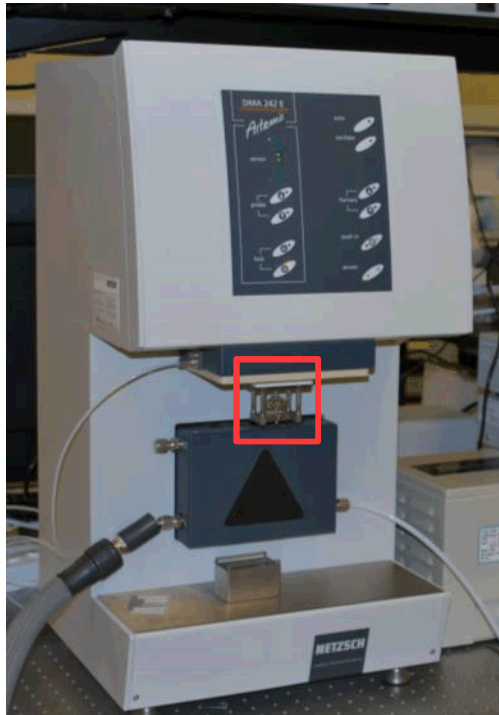
Cathode 94wt% LiCoO₂
3wt% PVDF
3wt% CB
void

binder



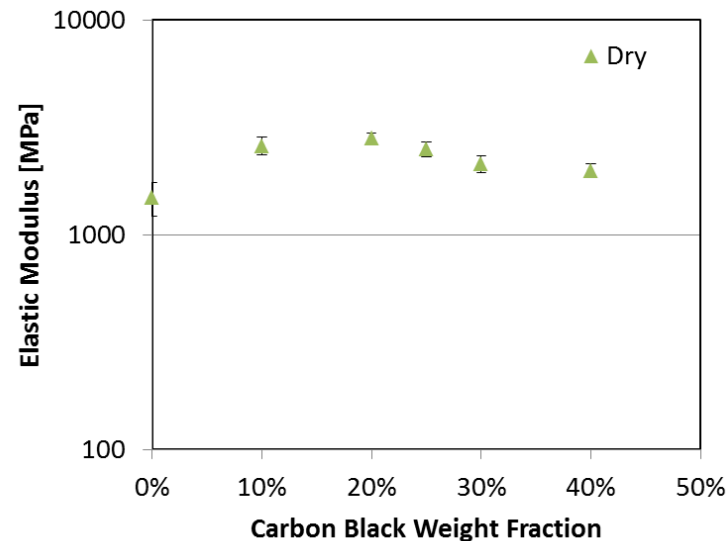
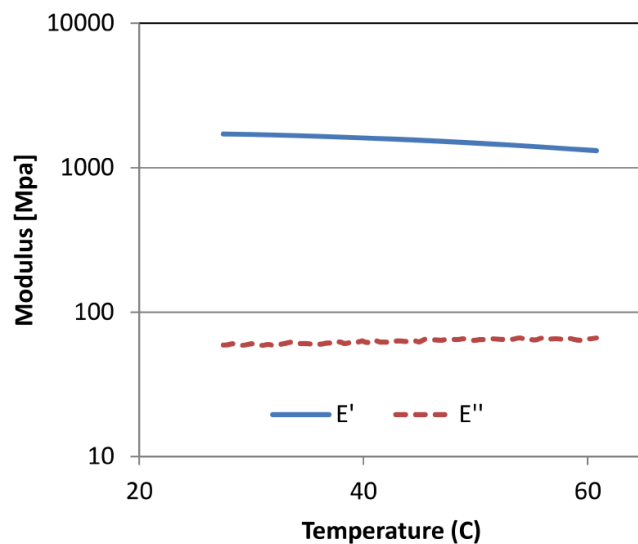
Anode 94wt% graphite
4wt% PVDF
2wt% CB
void

Mechanical Testing - DMA



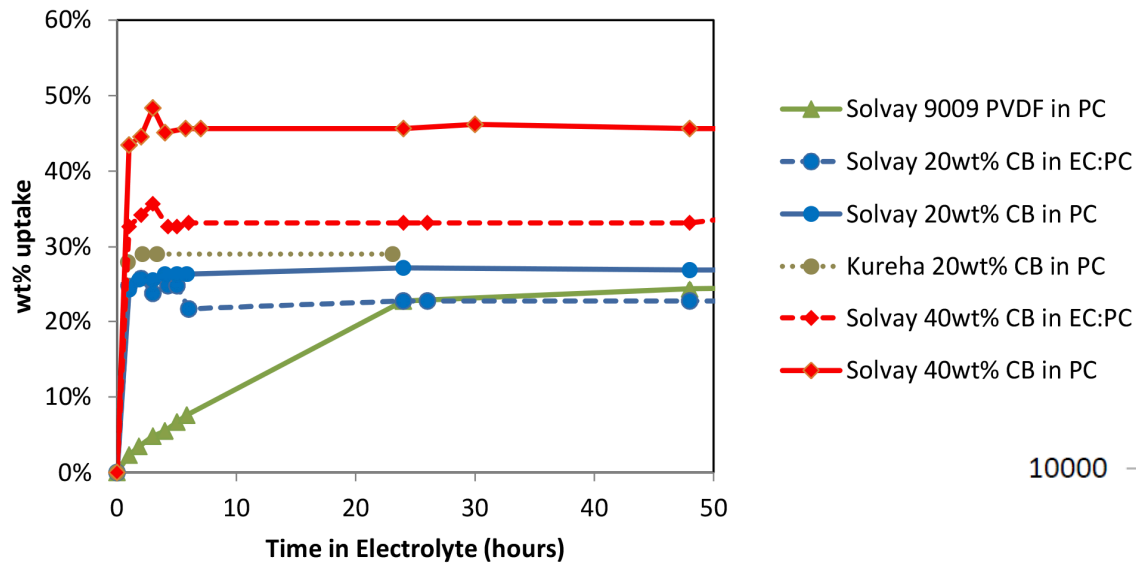
- Netzsch Artemis Dynamic Mechanical Analyzer
 - Probe E' , E'' from 1 – 10 Hz from 25 – 65 °C for dry samples
 - Probe E' , E'' from 1 – 50 Hz at room temperature for electrolyte (propylene carbonate) immersed samples

Mechanical Properties – Dry Binder



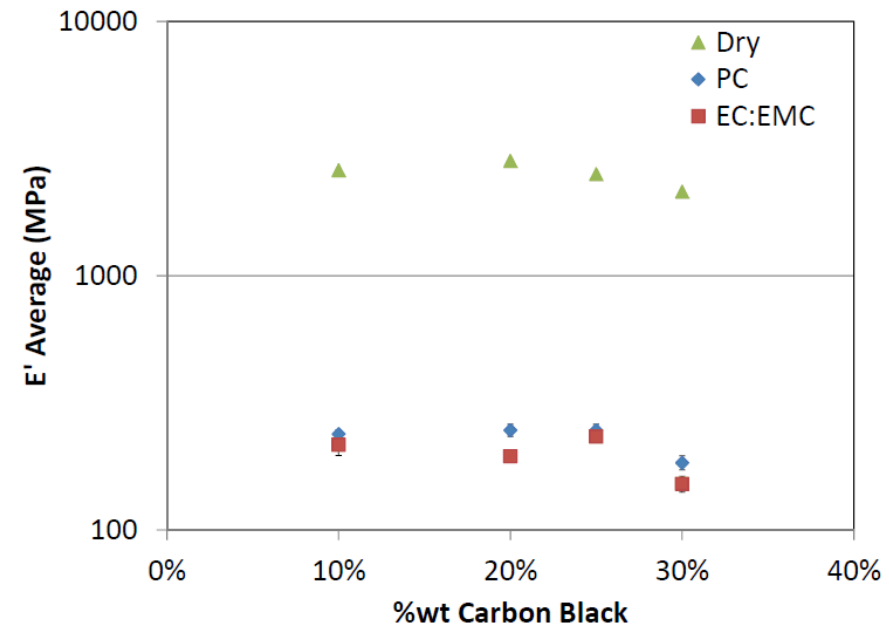
- Storage modulus >> Loss modulus for entire temperature range
- Binder softens at higher temperature
- Despite large differences in morphology, no dependence on carbon black concentration
- Elastic modulus for dry binder ranges between 1.5 – 2.8 GPa

Solvent Swollen (Wet) Binder Properties

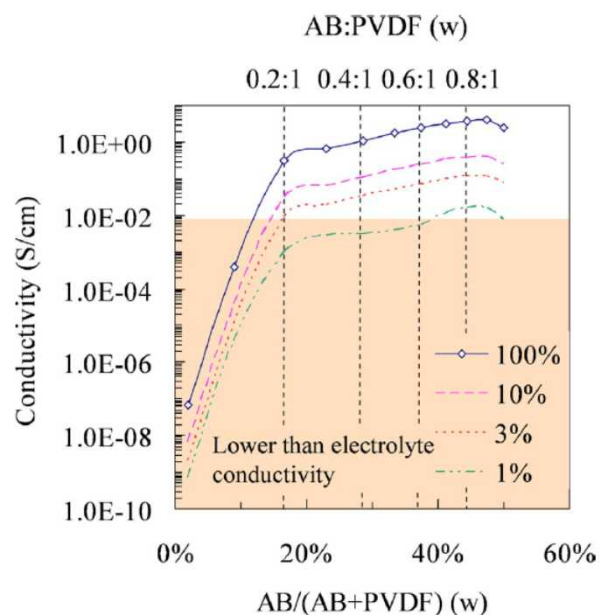
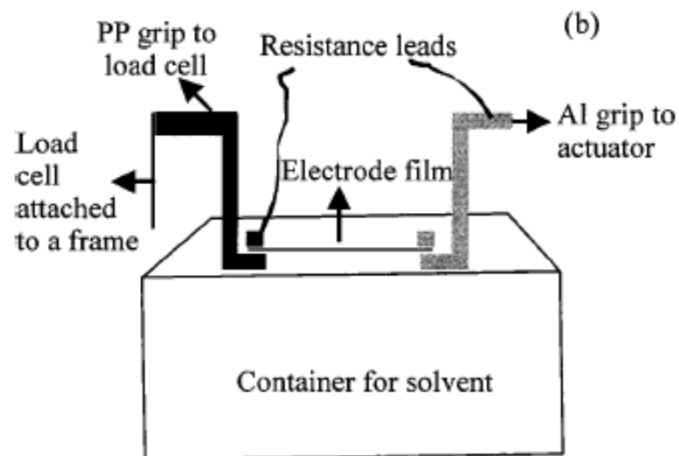


- Decreased modulus (≈ 200 MPa for wet compared to ≈ 2.4 GPa for dry)
- Peak modulus around 20wt% carbon black

- Immersed binder samples within propylene carbonate (PC) for at least 24 hours
 - Samples absorb between 25 – 45 wt% in propylene carbonate within 2 hours of immersion



Binder Conductivity in Literature

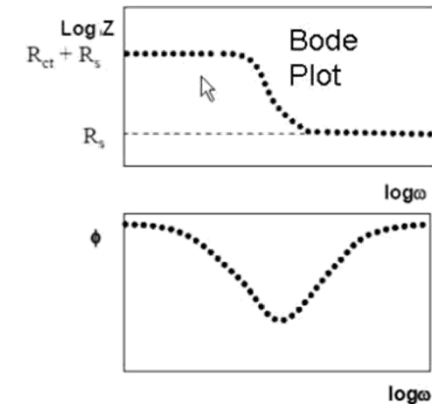
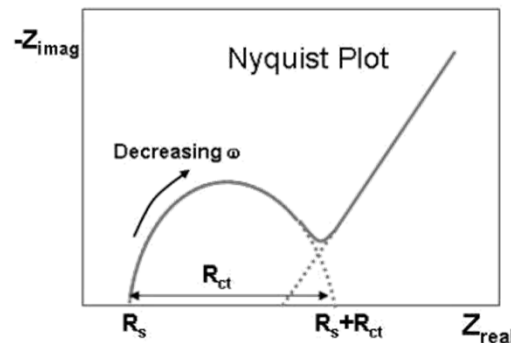
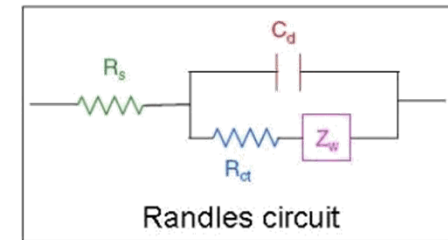
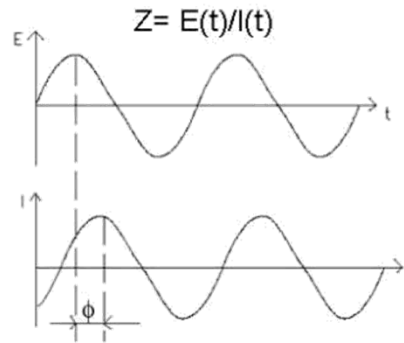


- Chen *et al.*¹ investigated the change in resistance of various PVDF/carbon composites (both wet and dry) and during initial cycling in tension
 - Small changes in resistivity with ~5 cycles number
- Liu *et al.*^{2,3} investigated the effect of varying carbon concentration on film conductivity
 - Substantial characterization of binder morphology, organization with varying CB concentration
- They both used 4 point probe technique to calculate conductivity
 - We are using oscillatory method
 - Measure different conductivities

Electrical Impedance Spectroscopy

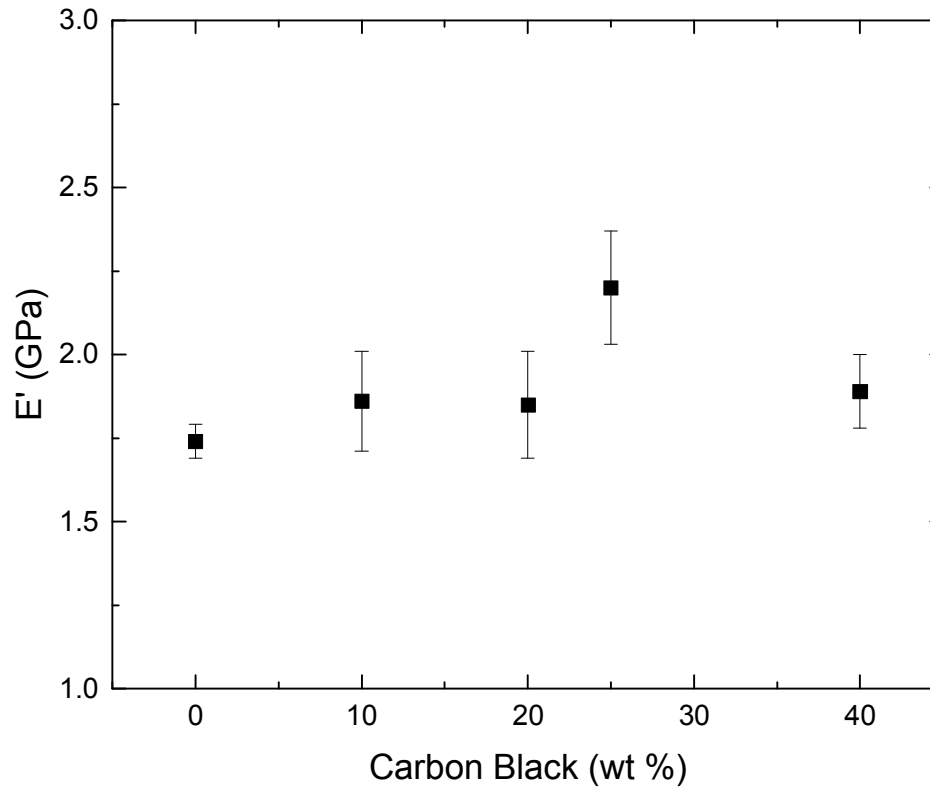
$$Z = \frac{E_t}{I_t}$$

$$Z = \frac{E \sin(\omega t)}{I \sin(\omega t + \phi)}$$



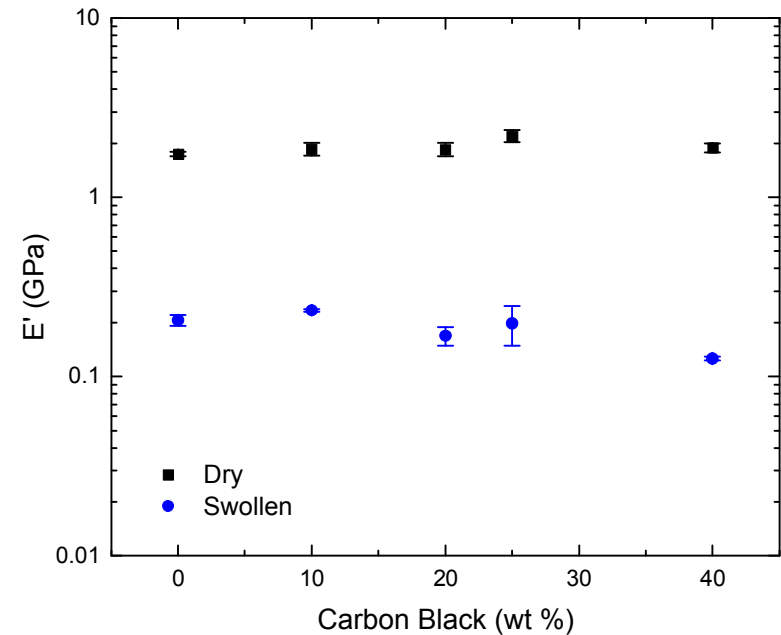
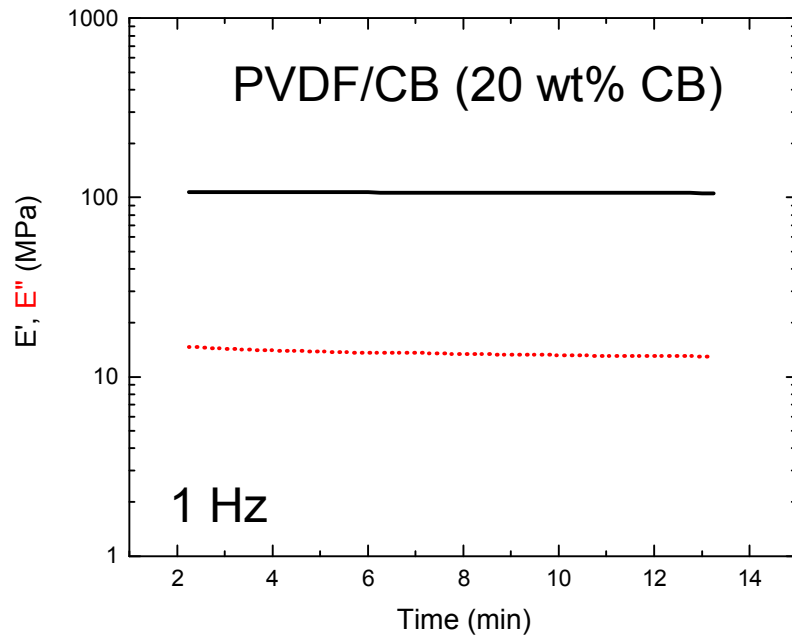
- AC equivalent to Ohm's Law
- Useful for characterizing electrochemical systems with various complex equivalent circuits (multiple resistances, capacitors, *etc.*)

Mechanical Properties – Dry Binder



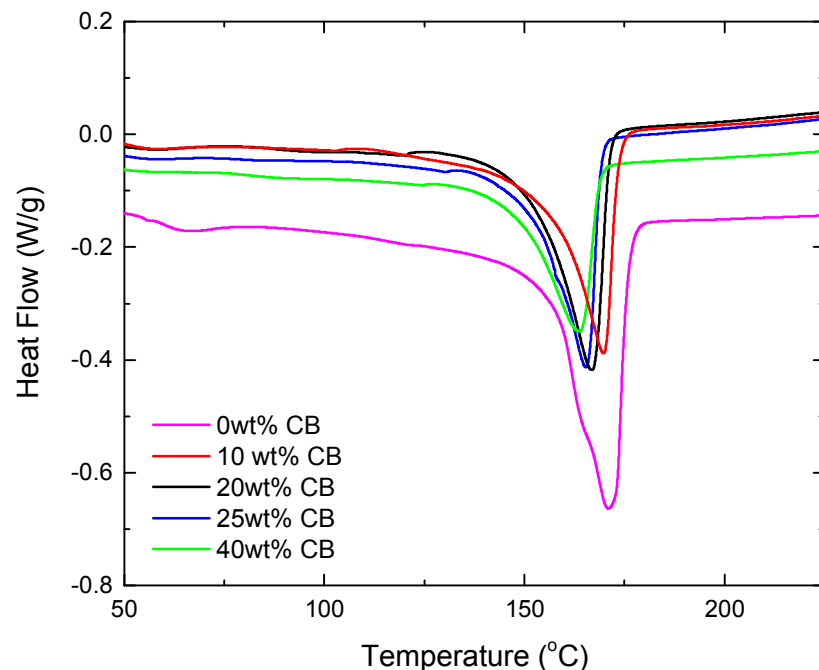
- No clear trend in modulus for varying CB weight percent
- Elastic modulus for dry binder ranges between 1.5 – 2.25 GPa
- Probe crystallinity with DSC to investigate microstructure of composites

Mechanical Properties - Swollen



- Immersed samples within propylene carbonate (PC) for at least 24 hours
 - Samples absorb between 25 – 35 wt% in propylene carbonate within 6 hours of immersion
- Decreased modulus (≈ 200 MPa for swollen compared to ≈ 2 GPa for dry)
- No clear trend as a function of carbon black weight percent

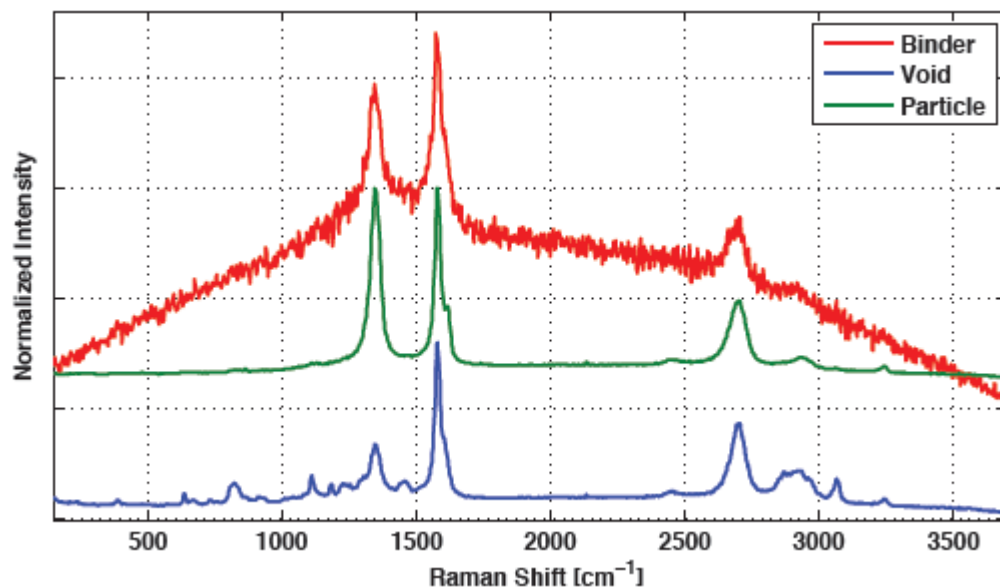
Differential Scanning Calorimetry



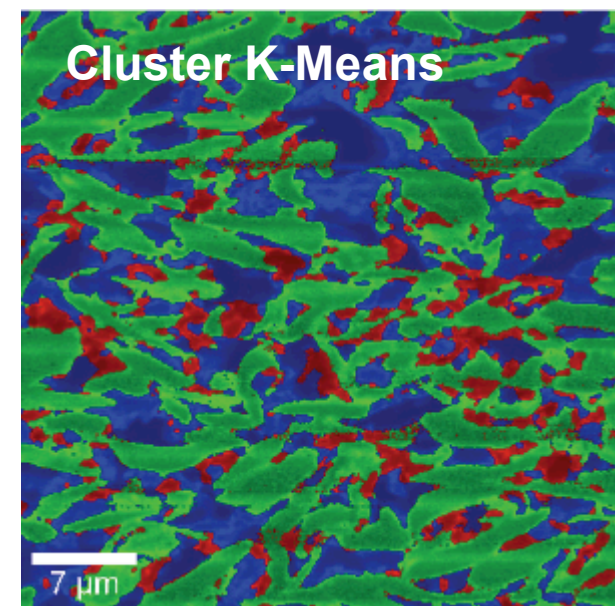
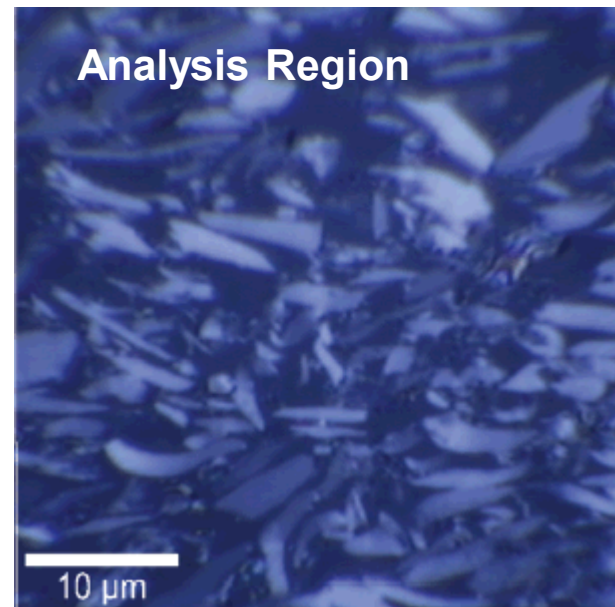
% CB	Normalized H_{sl} (J/g)	% Crystallinity
0.0	47.0	0.45
10.0	43.2	0.41
20.0	38.3	0.37
25.0	41.7	0.40
40.0	44.5	0.43

- Investigated % crystallinity using differential scanning calorimetry by quantifying latent heat of fusion and compared against fully crystalline PVDF ($H_{sl} = 104.7 \text{ J/g}$)¹
 - Adding carbon black does not change PVDF crystallinity
 - Hypothesize that crystalline PVDF structure controlling mechanical properties

- How to get the microstructure of the anode?
 - Graphite : PVDF/carbon black : epoxy (void)
 - Chemical similarity foils traditional methods

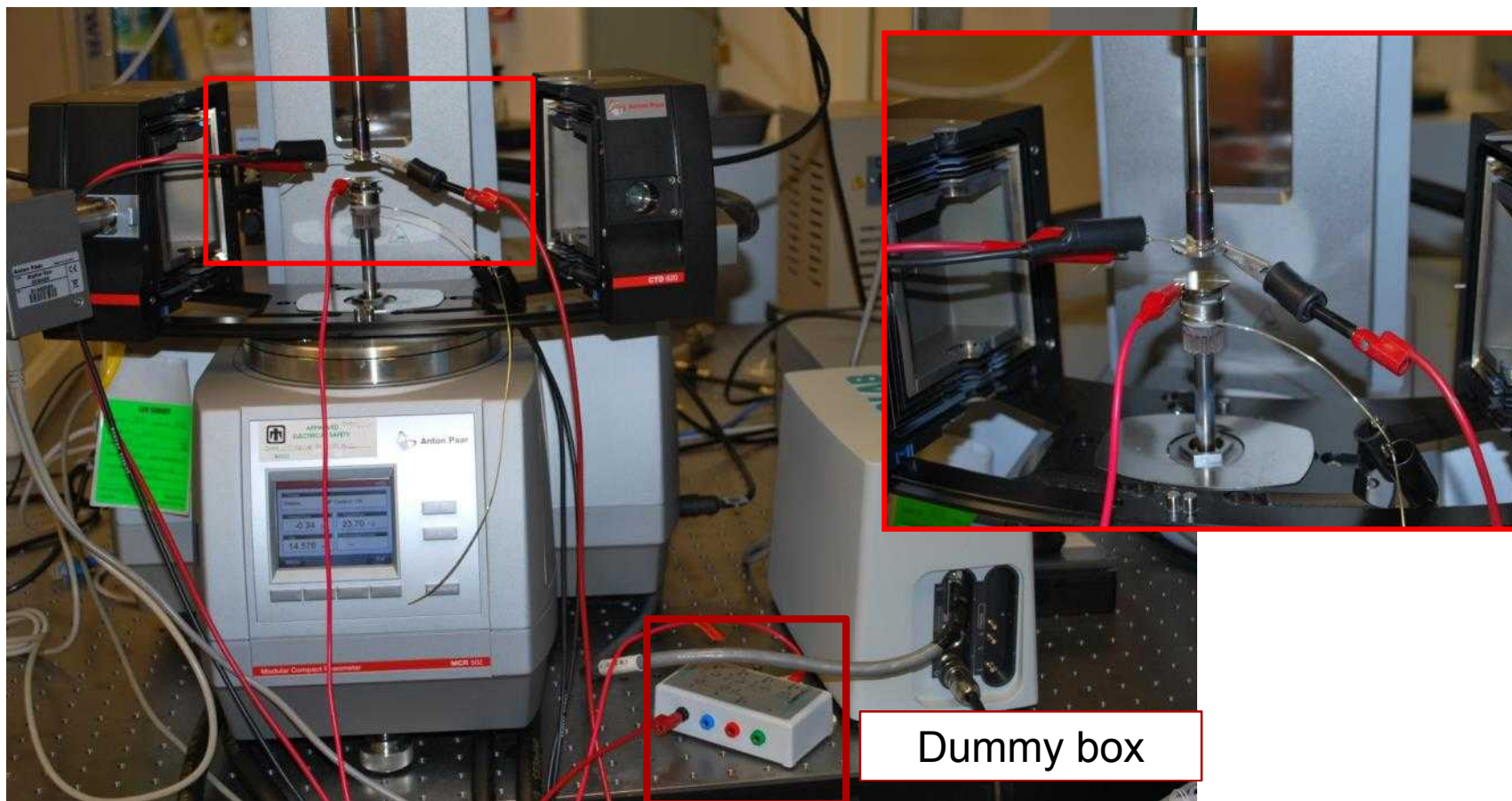


First mesoscale binder structures for an anode!
Raman may also be able to distinguish local states
of charge in graphite particles

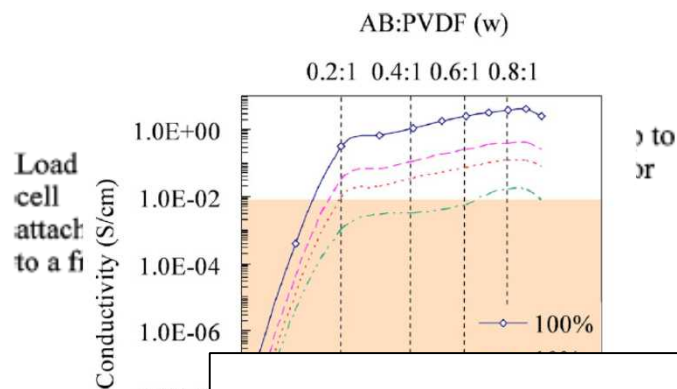


Testing contact/system/lead resistance

1. Ran potentiostat into dummy box (100 ohm resistor + (1000 ohm resistor in parallel with 1 uF capacitor)
2. Connected alligator clips into system, hooked into same dummy box connections

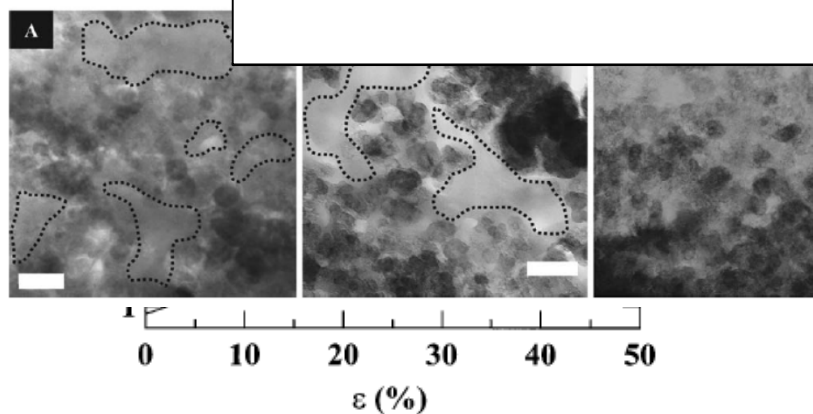


Binder Conductivity in Literature



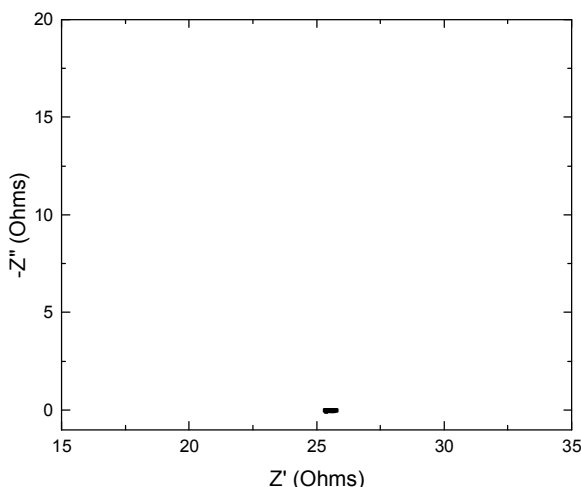
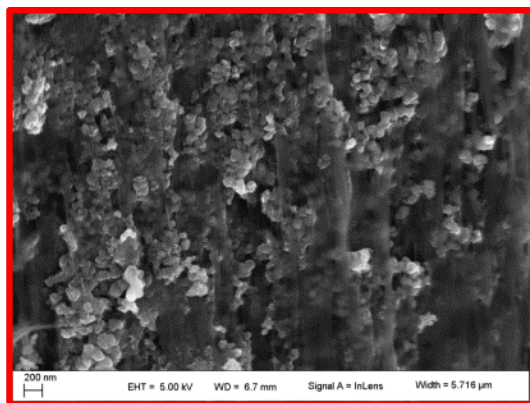
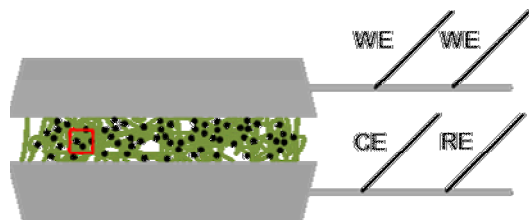
- Chen *et al.*¹ investigated the change in resistance of various PVDF/carbon composites (both swollen and dry) and during initial cycling in tension

Probe effects of binder conductivity as a function of mechanical compressive cycling



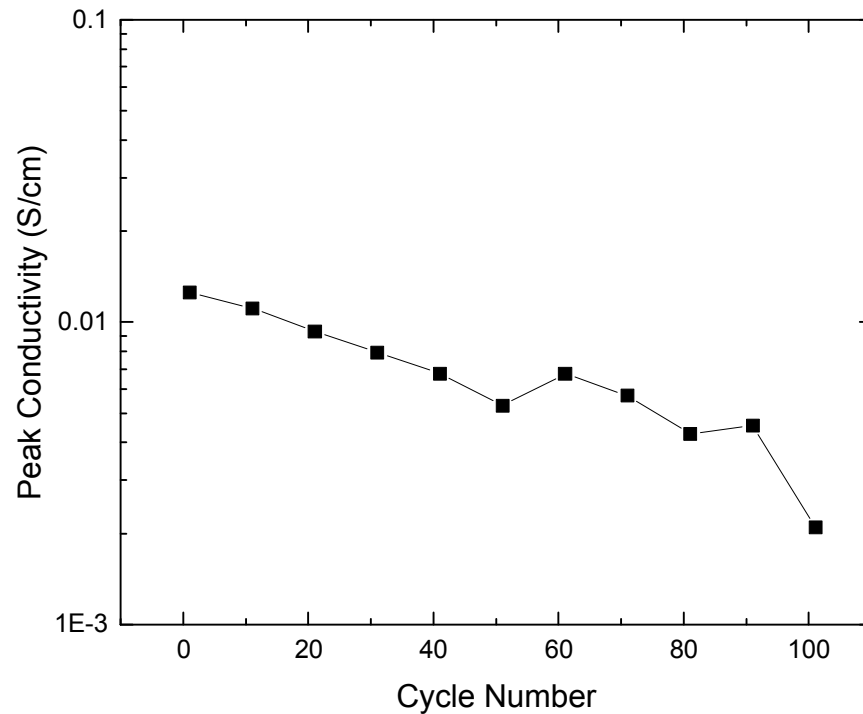
- Developed models to predict binder conductivity
- Substantial characterization of binder morphology, organization with varying CB concentration

Measurements on Dry Binder



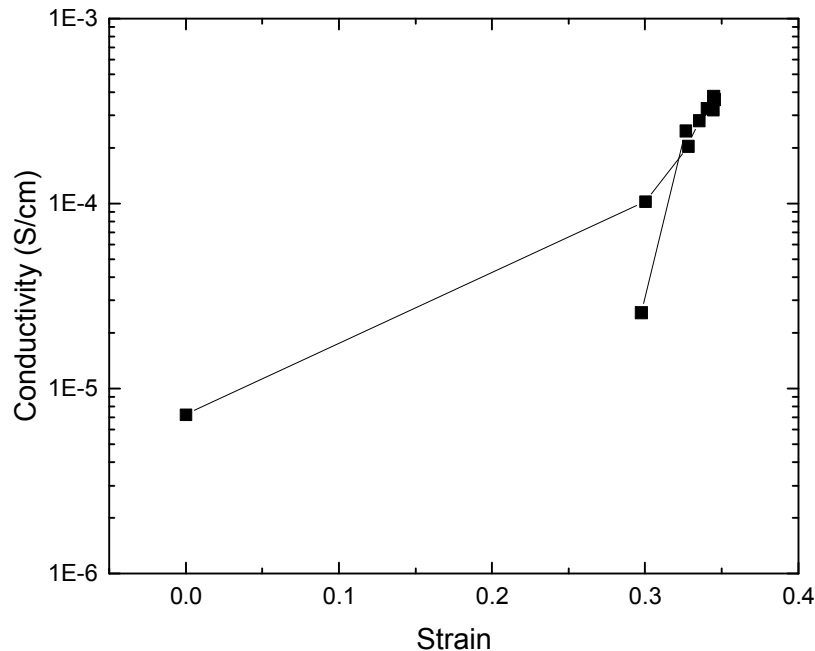
- Frequency response shows purely resistive behavior
 - No imaginary impedance (current measurement is completely in phase with applied voltage)
 - ****(easy system to characterize)****
- Using film dimensions, values can be converted to a conductivity
- Investigate conductivity as a function of the applied load and mechanical cycle

Cycling Behavior of Dry Composite Binder



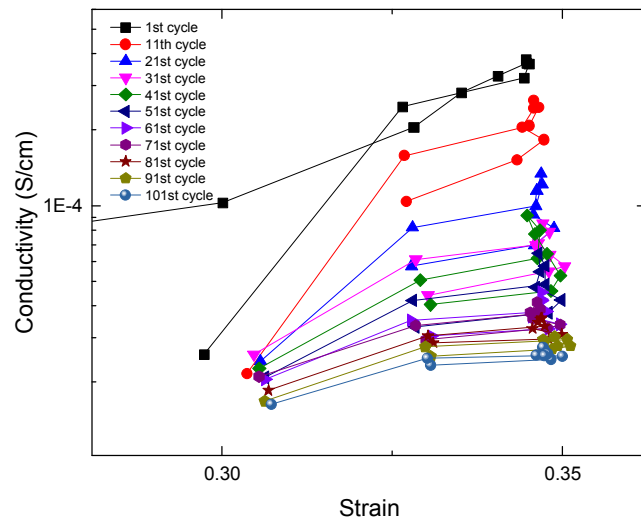
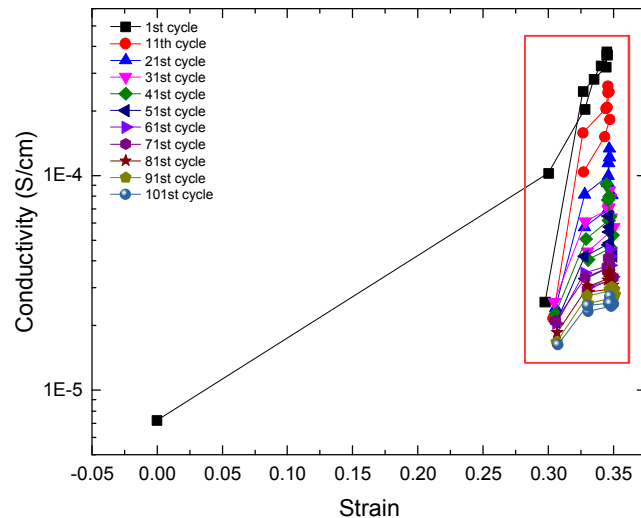
- Conductivity at 2.5 MPa (peak load/stress) as a function of cycle number
- PVDFCB – 40 wt%

Cycling Behavior of LCO Cathode



- I do not know the specifics of the cathode that we used other than:
 - 94 wt% LiCoO_2
 - 3 wt% PVDF
 - 3 wt% Denka CB
- Cathode exhibits a similar trend in increasing conductivity as a function of applied compressive stress
- Peak conductivity of cathode is almost two orders of magnitude lower than comparable binder
 - (40wt% vs. 50wt%)

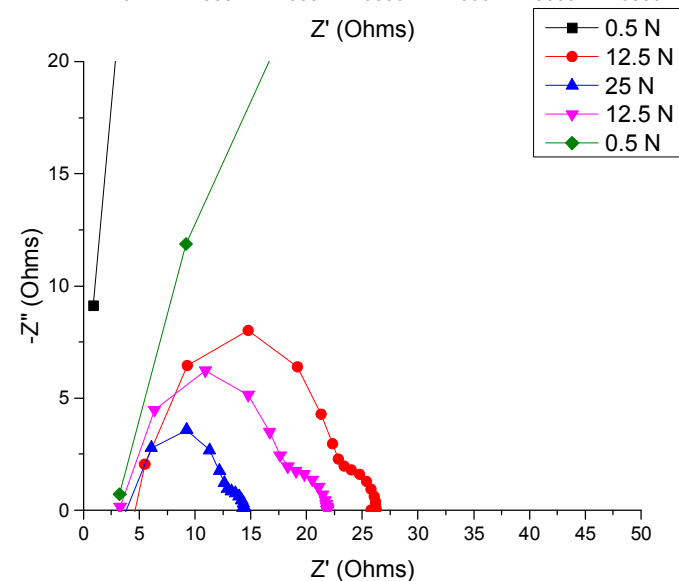
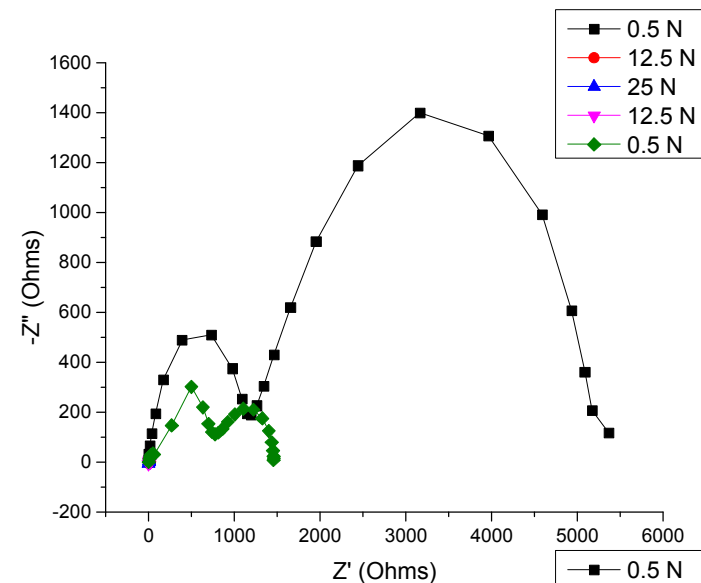
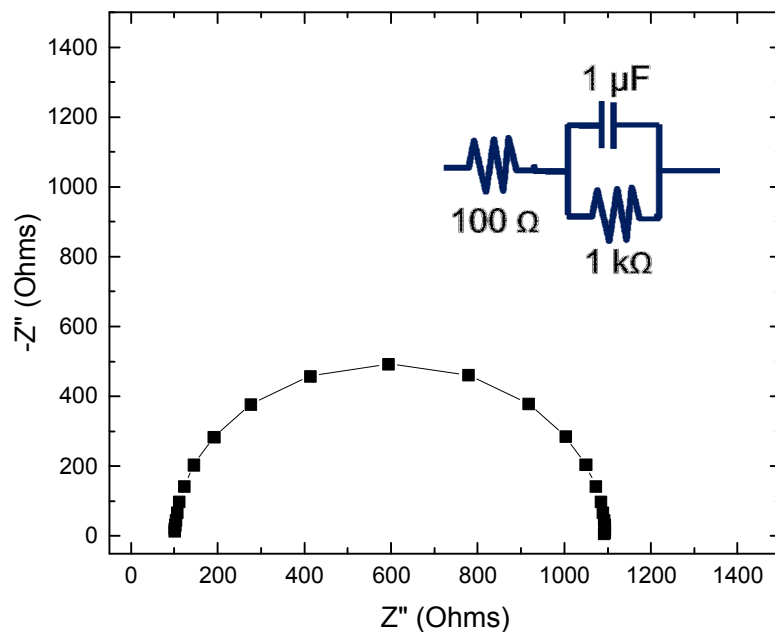
Cycling Behavior of LCO Cathode



- Procedure:
 - 250 kPa – 2.5 MPa – 250 kPa
 - 5C (cycle \approx 24 min), 9x 120C (cycle \approx 1 min)
 - 101 total cycles
- Just as with the binder, the conductivity as a function of cycling drops substantially (greater than an order of magnitude)
- (just a note)
 - Interesting that the conductivity doesn't drop dramatically from the first cycle to the 11th (after the apparently huge plastic deformation)

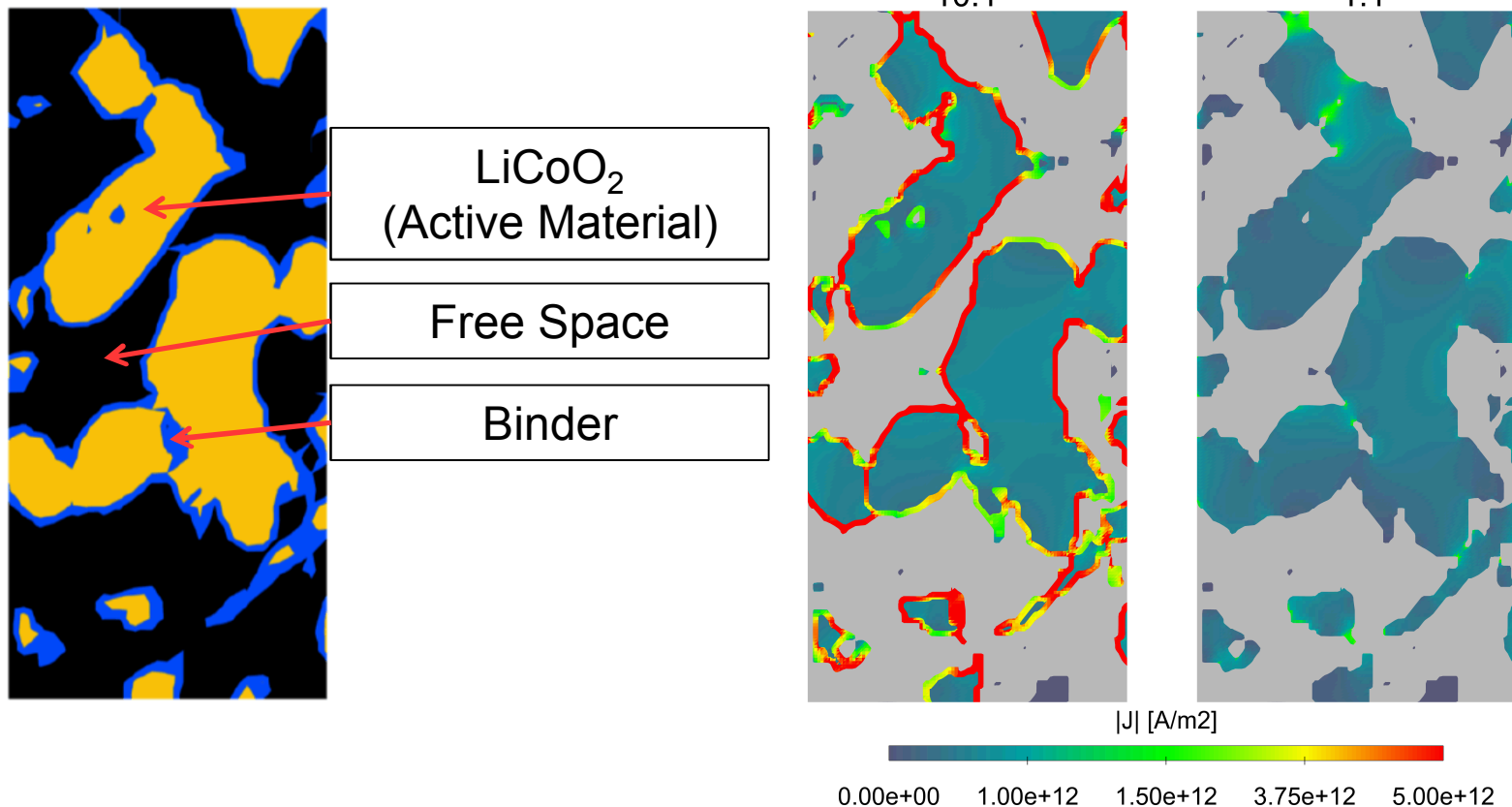
Wet cathode Electrical Impedance Spectroscopy

- 1MHz – 50 mHz, 10 mV
- Possibly crushed swollen cathode?



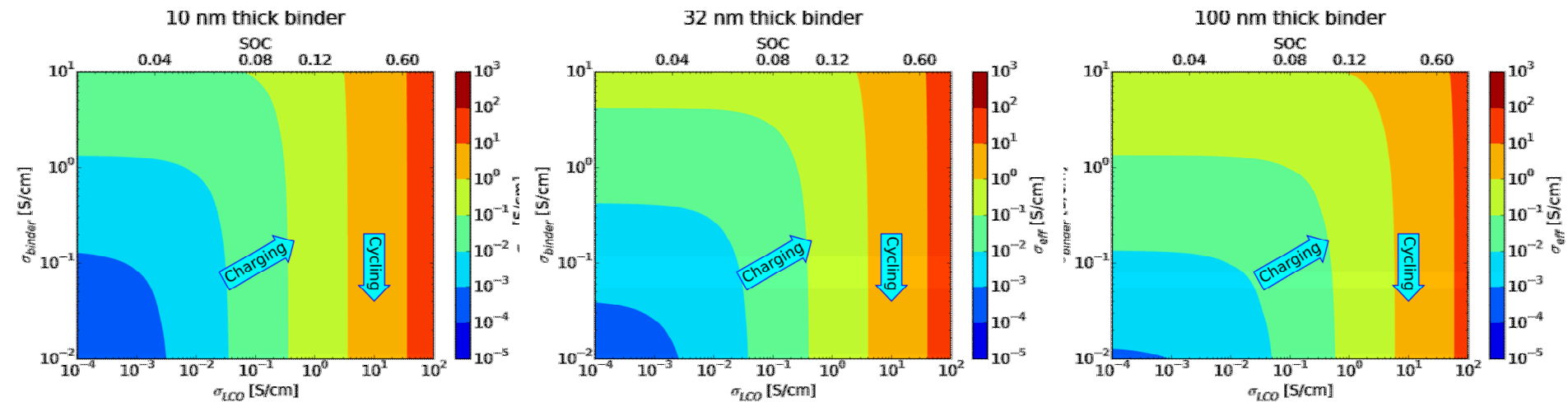
Impact of Binder Conductivity on Cathode

- Calculate the effective electrical conductivity of a representative cathode microstructure as the binder conductivity decreases.



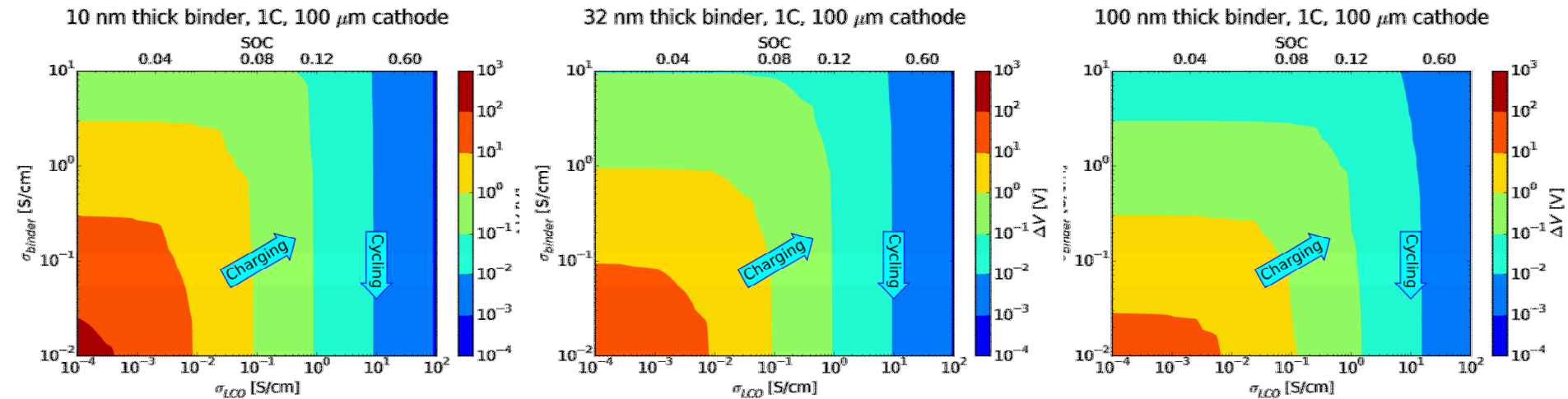
Effective conductivity

- Effective conductivity of the network with 3 binder thicknesses
- Conductivity increases during charging
 - LCO conductivity increases
 - Binder becomes stress and increases conductivity
- With cycling, network conductivity decreases (but only at low SOC)
 - Binder conductivity decreases with cycling
 - Most apparent for low SOC and thick binder coatings
- Is this a potential mechanism of capacity fade; losing the last 5-10% of capacity due to low effective conductivity?



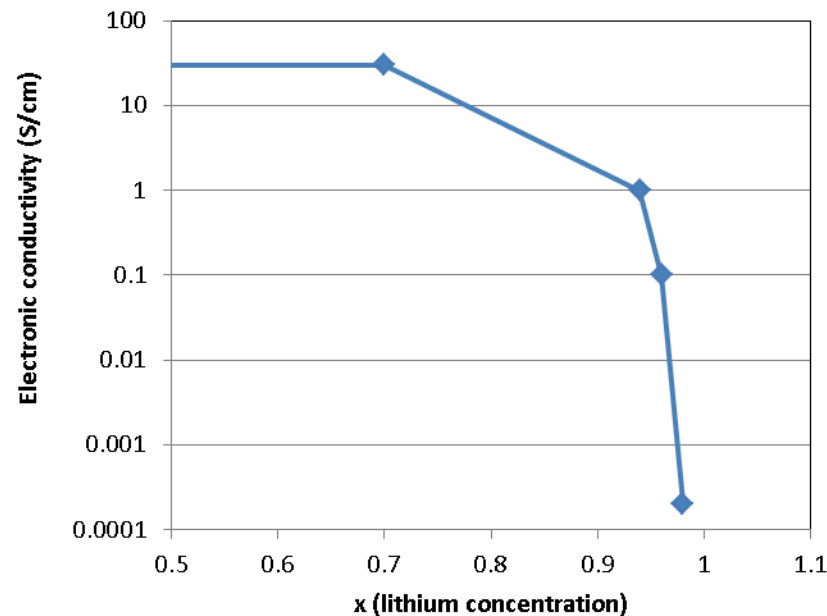
Voltage drop

- Another way to look at this is in terms of the voltage drop across the cathode
 - 100 μm cathode, 1C (dis)charge rate
- At high SOC, there is very little voltage drop ($< 0.1\text{V}$ for $\text{SOC} > 0.1$)
- For the last 5% SOC (discharging), the binder conductivity becomes important
 - Binder conductivity degradation from cycling could significantly increase voltage drop, into the 1-10 V range (rendering battery useless)
- Voltage drop scales linearly with current, so this is exacerbated for higher discharge rates



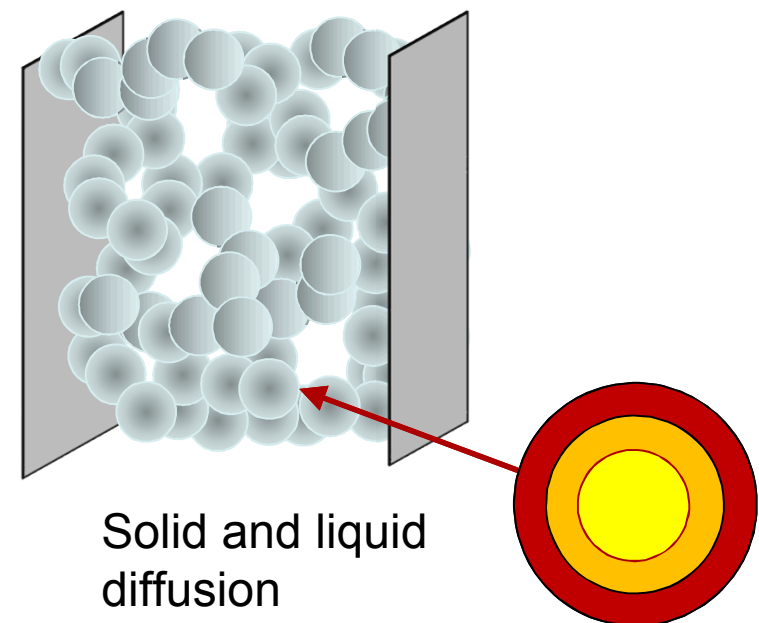
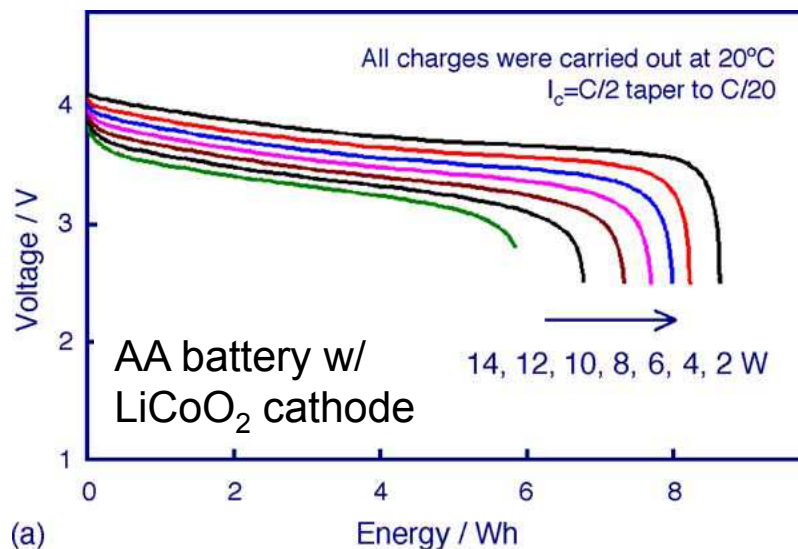
Electronic Conductivity of LiCoO₂

- Electronic conductivity as a function of degree of lithiation shows an insulator to metal transition during deintercalation of lithium.



High Power Draw Limitations

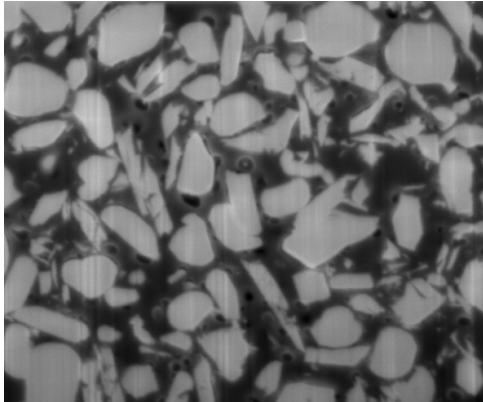
- Voltage drops caused by battery internal resistance
 - Transport limitations – ohmic losses, lithium ion depletion
 - Reaction rate limitations



- Part of a larger effort at Sandia to develop predictive models for battery performance
 - Electrochemical, mechanical, and thermal dependence

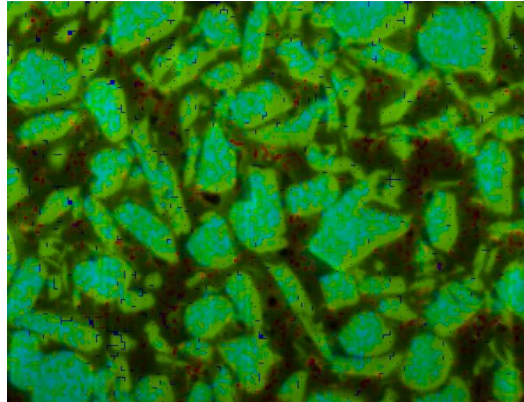
Location of Binder

Focused Ion Beam
Cross-section

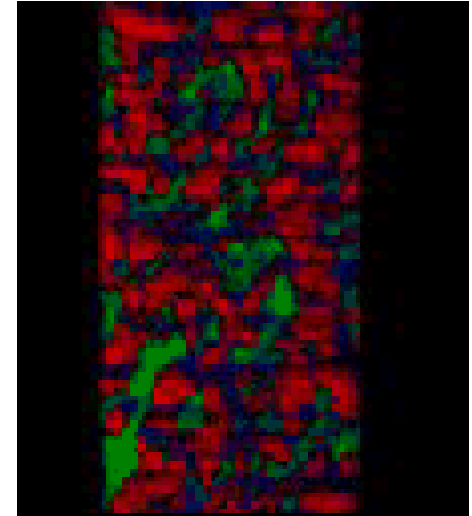


LiCoO₂ particles
55 vol% active
material
mean particles size
2.3microns
Aspect ratio of 1.7

Energy Dispersive Spectroscopy
chemical composition



Multivariate principal
component analysis



Red – LiCoO₂
Blue – PVDF/CB binder
Green - void

Binder found in the small spaces between particles

Tortuosity, Porosity and Network Conductivity

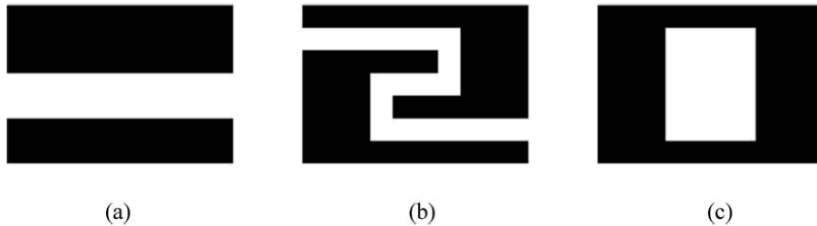
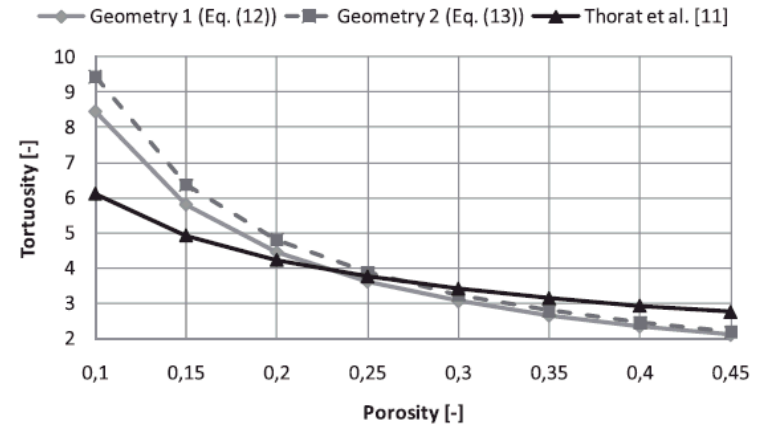


Figure 2. Three 2D porous media with a porosity of $2/7$ and tortuosities of 1 (a), 2 (b), and infinity (c). While (a) and (b) show open pores, (c) displays a closed pore that does not contribute at all to material transport.

Kehrwald et al. (2011)



Rubino et al. (2001)

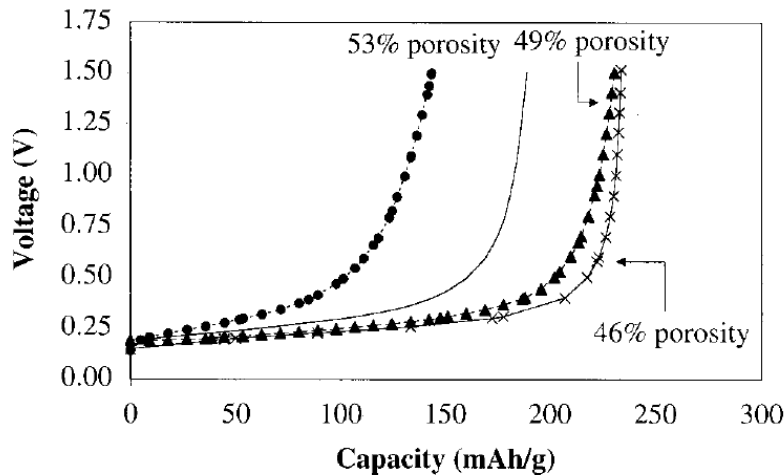
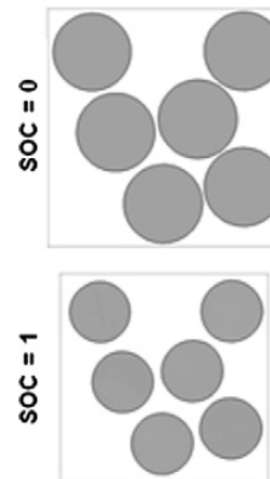
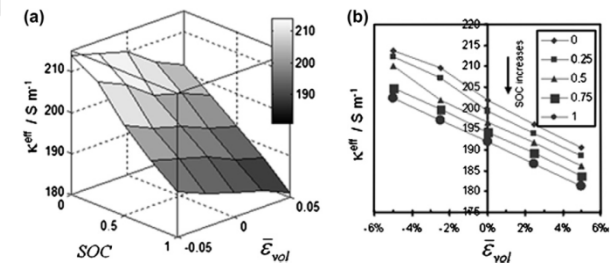
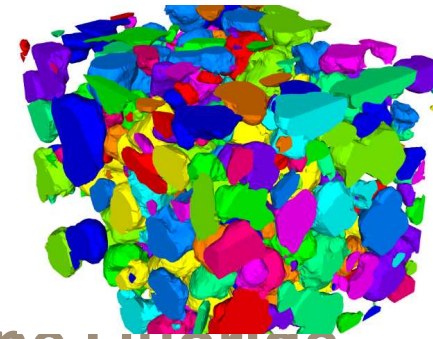
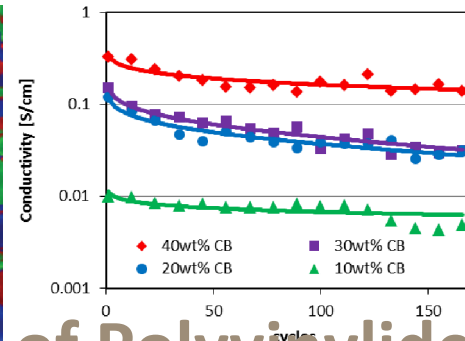
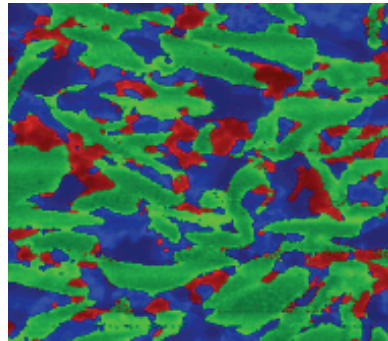
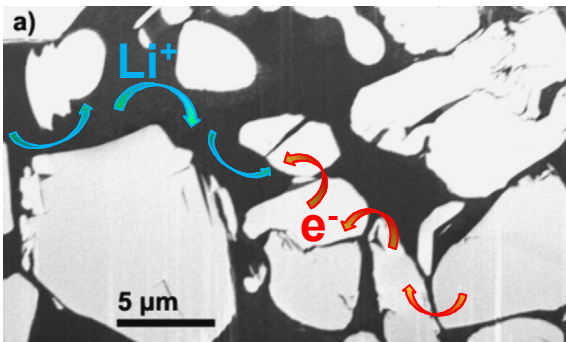


Figure 4. Half-cell voltage curves for anode pieces recovered from cycled cells. Half-cells were discharged to 0.010 V at $C/2$ followed by a constant potential step with a $C/20$ current cutoff. Voltage curves shown are for a $1C$ charge to 1.50 V. Circles: prismatic cell (300 cycles), line: prismatic cell (300 cycles, recompact), triangles: cylindrical cell (300 cycles), crosses: prismatic cell (formation only). Curves represent the best cell of two.



Awarke et al. (2011)
Network conductivity





Conductivity Degradation of Polyvinylidene fluoride Binder During Cycling: Measurements and Simulations for Lithium Ion Batteries

Anne M. Grillet, Thomas Humplik, Emily K. Stirrup, David A. Barringer, Scott A. Roberts, Chelsea M. Snyder, Madison R. Janvrin & Chris A. Apblett

June 2, 2016 #A02-0368

229th ECS Meeting



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