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# Predicting Strain in Cure Shrinkage Induced Epoxy/Metal Bilayer Beam Bending

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# Cure Shrinkage Induced Bending in Epoxy/Metal Bilayer Material

- Thin layer of metal coated with thermoset epoxy



- During cure, epoxy shrinks causing bilayer to bend
  - Can cause component failure, interface cracking, etc.

# SEMI-ANALYTIC MECHANICS THEORY

# Timoshenko's Formulae for Eigen Strain

## Induced Beam Bending

- Classic 1925 paper focuses on bilayer metallic beams bending due to thermal expansion mismatch
- Same concept applicable to other eigen strains
  - Cure shrinkage
- Formula for radius of curvature ( $\rho$ )

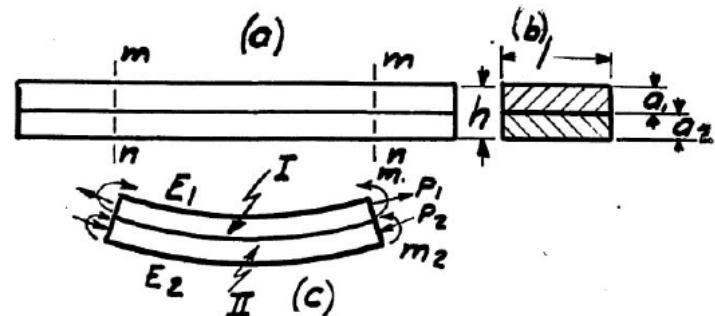


FIG. 1. Deflection of a bi-metal strip while uniformly heated.

$$\frac{1}{\rho} = \frac{6\varepsilon_\Delta}{h \left( 3 + (1 + mn^3) \left( 1 + \frac{1}{mn} \right) / (n + 1)^2 \right)}$$

- Where,  $\varepsilon_\Delta$  is the eigen strain mismatch,  $m = \frac{E_1}{E_2}$  is the ratio of elastic moduli,  $n = \frac{a_1}{a_2}$  is the ratio of thicknesses, and  $h = a_1 + a_2$  is the total thickness

# Predicting Strain at Bottom of Bilayer Beam

- The strain at the bottom of a beam in bending,

$$\varepsilon_{bottom} = \alpha_2 \Delta T + \frac{F_2}{E_2 a_2} + \frac{a_2}{2\rho}$$

where,

- $F_2$  is the axial force placed on the second layer by the first

Simplified using,  $F_2 = \frac{2(E_1 I_1 + E_2 I_2)}{h\rho}$  and the bilayer

ratios,  $m = \frac{E_1}{E_2}$ ,  $n = \frac{a_1}{a_2}$

$$\varepsilon_{bottom} = \alpha_2 \Delta T + \frac{a_2}{\rho} \left[ \frac{mn^3 + 3n + 4}{6(n + 1)} \right]$$

# Dependence of Bending on Geometric and Material Properties during Cure

- Timoshenko's formula depends on properties that are **not constant** during the curing process
  - $m = \hat{m}(T, x)$ , **modulus of epoxy** varies with temperature and reaction extent ( $x$ )
  - $n = \hat{n}(T, x)$ ,  $h = \hat{h}(T, x)$ , geometry can vary with temperature and reaction extent – likely negligible
  - $\varepsilon_\Delta = \hat{\varepsilon}_\Delta(T, x)$ , eigen strain mismatch (**cure shrinkage**) varies with temperature and reaction extent – Important!
- Variation of moduli and cure shrinkage with temperature and reaction extent **captured by experiments**
  - SAND2013-8681

# Evolution of Epoxy Shear Modulus during Cure

- Modeled as,

$$\bar{G} = \left( 1 + \frac{\partial \bar{G}}{\partial T} (T - T_{ref}) \right) f(x)$$

where,

- $\bar{G} = \frac{G_\infty}{G_{\infty_{ref}}}$  is the normalized equilibrium shear modulus where  $G_{\infty_{ref}}$  corresponds to  $T_{ref}$
- $f(x) = \left[ \frac{x^2 - x_{gel}^2}{x_{ref}^2 - x_{gel}^2} \right]^{\frac{8}{3}}$ , and  $x_{gel}$  is the reaction extent at gel, and  $x_{ref} = 1$

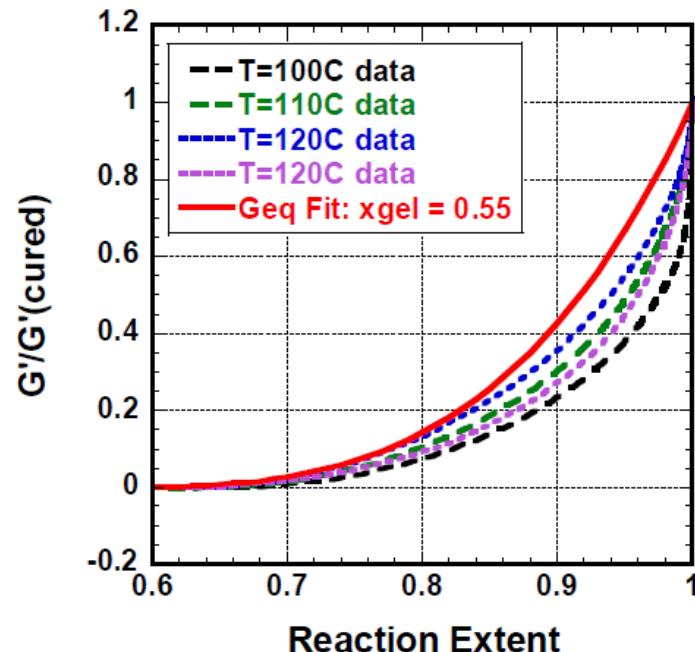


Figure 3-13. Parameterization of Equation 17 to measurements of equilibrium shear modulus as a function of reaction extent.

Source: SAND2013-8681

# Modulus Ratio Function of Temperature and Reaction Extent

- The modulus ratio function  $\hat{m}$  is,

$$\hat{m}(T, x) = \frac{E_{1_{ref}}}{E_2} \bar{G}(T, x),$$

where  $E_{1_{ref}} = 2G_{\infty_{ref}}(1 + \nu_1)$

# Variation of Cure Shrinkage over Reaction Extent with Temperature

- Cure shrinkage represented by volumetric strain is approximately linear after the gel point
  - $\varepsilon_{cure} = \frac{\beta_\infty}{3} (x - x_{gel}) \mid x > x_{gel}$
  - $0.060 < \beta_\infty < 0.082$
- Open questions
  - Should  $\beta_\infty$  be a function of temperature also?

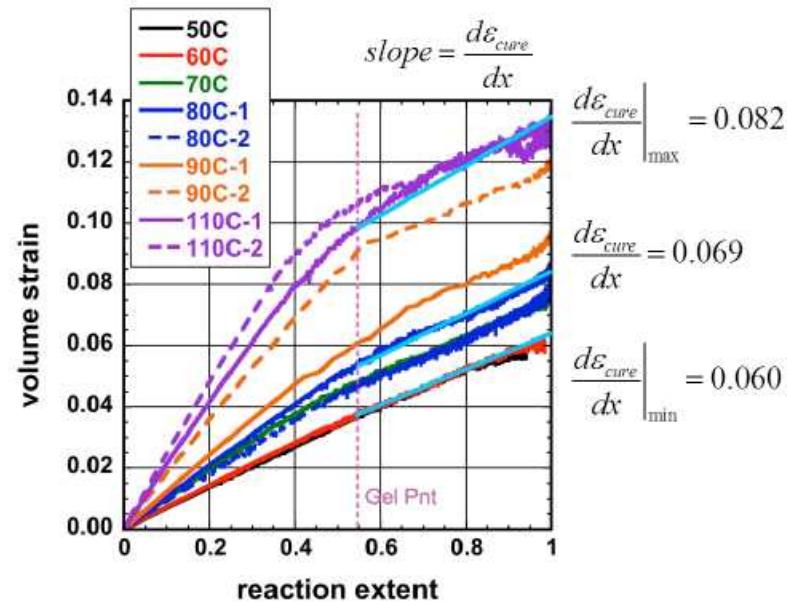


Figure 3-9. The volume strain associated with cure shrinkage versus the reaction extent. Volume shrinkage is determined based on the extrapolated polymer densities in Figure 3-7 using Equation 15. Reaction extent is determined from Equation 7 with fitting parameters described in Table 3-1 and  $w=0$ . This does not account for (1) non-isothermal conditions at early times or (2) vitrification during reaction. The 1<sup>st</sup> derivative beyond the gel point is evaluated for some datasets.

Source: SAND2013-8681

# Expression of the Eigen Strain Mismatch with Cure Shrinkage and Thermomechanical Strain

- The eigen strain mismatch is the combination of the strains associated with curing and temperature

$$\varepsilon_{\Delta} = \varepsilon_{cure} + \alpha_2 \Delta T - (\alpha_1 \Delta T - \varepsilon_{ref})$$

$$\hat{\varepsilon}_{\Delta}(T, x) = \frac{\beta_{\infty}}{3} (x - x_{gel}) + \alpha_2 \Delta T - (\alpha_1 \Delta T - \varepsilon_{ref}),$$

when  $x \geq x_{gel}$

- The cure strain represents a volumetric shrinkage, i.e.  $\boldsymbol{\varepsilon} = -\varepsilon_{cure} \mathbf{1}$ 
  - As it is fit as a positive magnitude, signs cancel leading to above form
- $\Delta T$  represents the temperature change from the gel point
- $\varepsilon_{ref}$  represents the change in stress free configuration – can vary through the thickness!

# Tracking the Change in Stress Free Configuration During Epoxy Cure

- The stress free configuration of the epoxy changes during the cure process due to the addition of cross links. Thus, the typical expression for thermal strain has an error.
- The offset from the original configuration is tracked through a reference strain such that,

$$\varepsilon_{ref}^* = \frac{1}{G_\infty(t_n)} \int_{t=0}^{t=t_n} G_\infty(t) \frac{d\varepsilon_{dev}}{dt} dt - \varepsilon_{dev}(t_n)$$

- We specifically care about the in-plane mechanical strain,

$$\varepsilon_{pp} = \frac{P}{E_1 a_1} = \frac{h}{6\rho} \left[ \frac{mn^3 + 1}{mn(n+1)^2} \right]$$

$$\varepsilon_{tt} = -\frac{2\varepsilon_{pp}\lambda}{2G_\infty + \lambda}, \lambda \text{ is the lame constant}$$

$$\varepsilon_{pp,dev} = \varepsilon_{pp} - \frac{1}{3} (2\varepsilon_{pp} + \varepsilon_{tt})$$

# Reaction Extent dependence on Temperature

- Rate of Reaction is dependent on current reaction extent ( $x$ ) and temperature

$$\frac{dx}{dt} = \hat{k}(b + x^m)(1 - x)^n$$
$$\hat{k} = \frac{k_0 \exp\left(-\frac{E_a}{RT}\right)}{(1 + wa)^\beta}$$

where,

- $b, m, n, w, \beta, k_0$  are fitted constants related to various aspects of the reaction process (See SAND2013-8681)
- $E_a$  is the activation energy
- $a$  represents a shift factor due to vitrification
- $\beta$  is set to 0 to ignore the shift effects.

# Strain in Bilayer Beam due to Cure Shrinkage Induced Bending

- Formulae for each different aspect have been discussed
  - Dependence of eigen strain mismatch on temperature and reaction extent
  - Dependence of epoxy shear modulus on reaction extent and temperature
  - Evolution of Stress Free Configuration with Modulus and Strain
  - Variation of rate of reaction (and reaction extent) on temperature
- The full solution requires:
  - Numerical integration of the rate of reaction to determine the reaction extent from the temperature data
  - Solution of evolving reference configuration and curvature
  - Calculation of strain at desired location
  - Comparison to experimental data

# Rate of Reaction Equation Solution

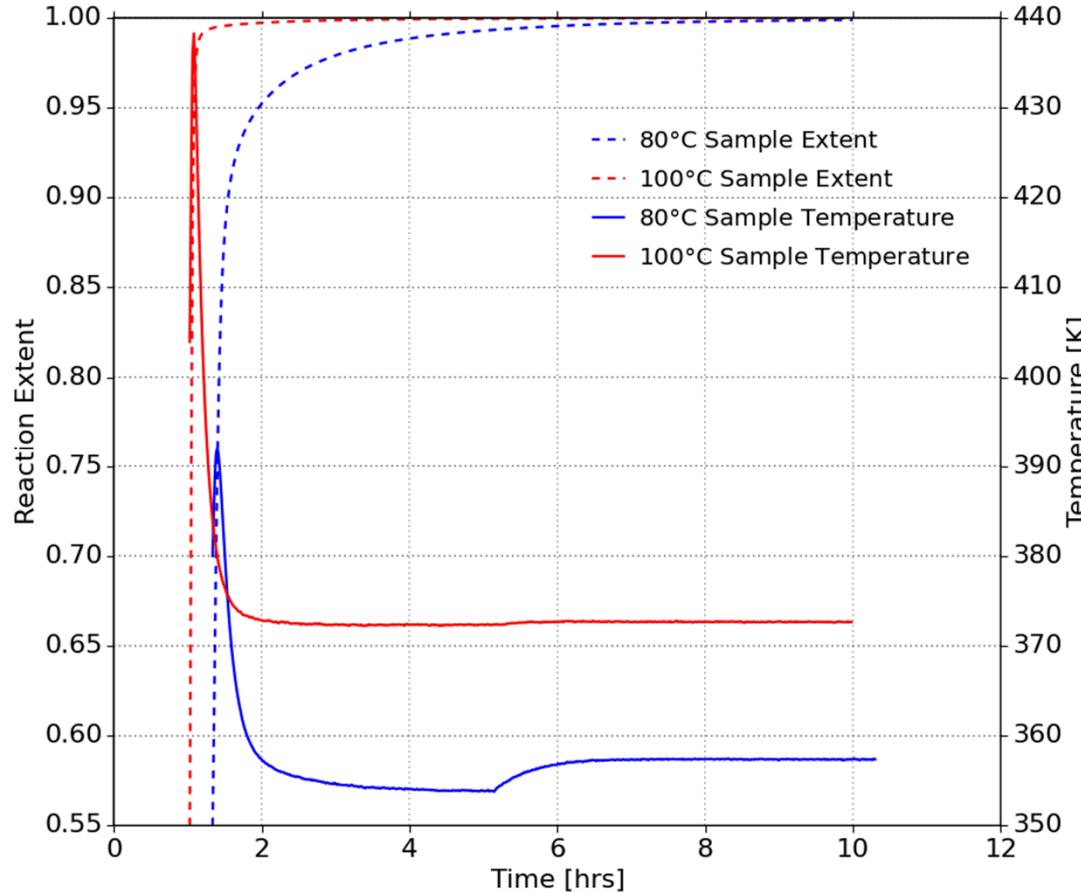
## Parameters

- Solution done in python
  - Scipy: odeint integrator
- Rate of Reaction parameter list (SAND2013-8681)

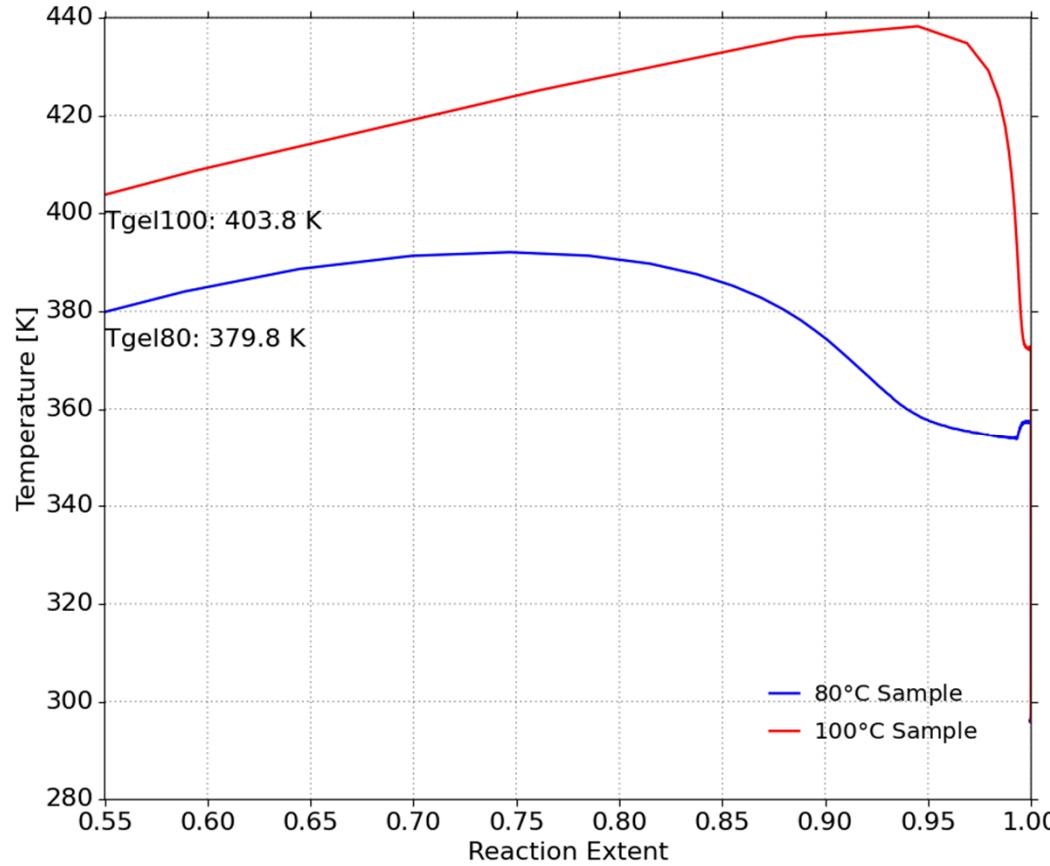
Parameter	Value
$E_a$	13.8 [kcal/mole]
$k_0$	217e+03 [ $s^{-1}$ ]
$b$	0.17
$m$	0.33
$n$	1.37
$w$	-
$a$	-
$\beta$	0

WLF Shift  
Parameters are  
unused

# Higher Cure Temperature Results in Higher Rate of Reaction



# Temperature of Experimental Samples throughout Reaction Extent

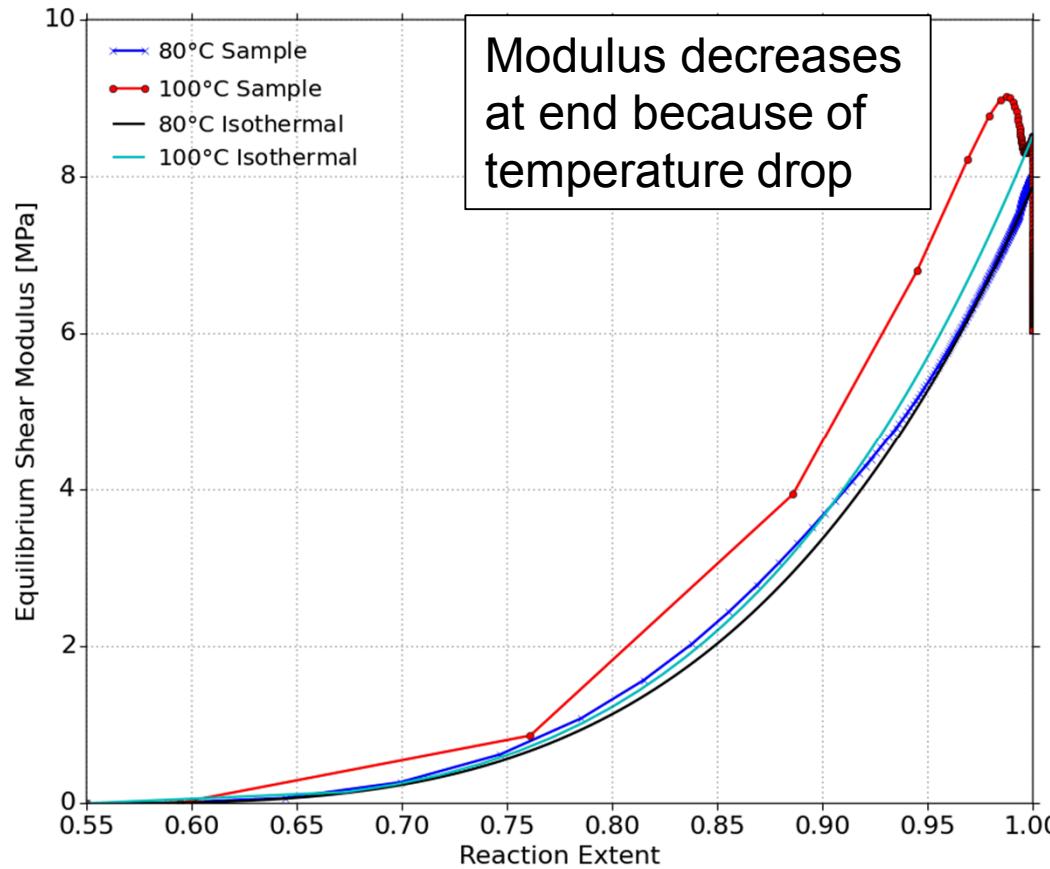


# Shear Modulus Evolution Parameters

- The shear modulus is calculated using the proposed model and the calculated reaction extent for each data point
- The model uses the following parameters (SAND2013-8681)

Parameter	Value
$x_{gel}$	0.55
$x_{ref}$	1
$T_{ref}$	90 [deg C]
$\partial \bar{G} / \partial T$	0.0039 [deg $K^{-1}$ ]

# Higher Cure Temperature results in Higher Modulus during Cure

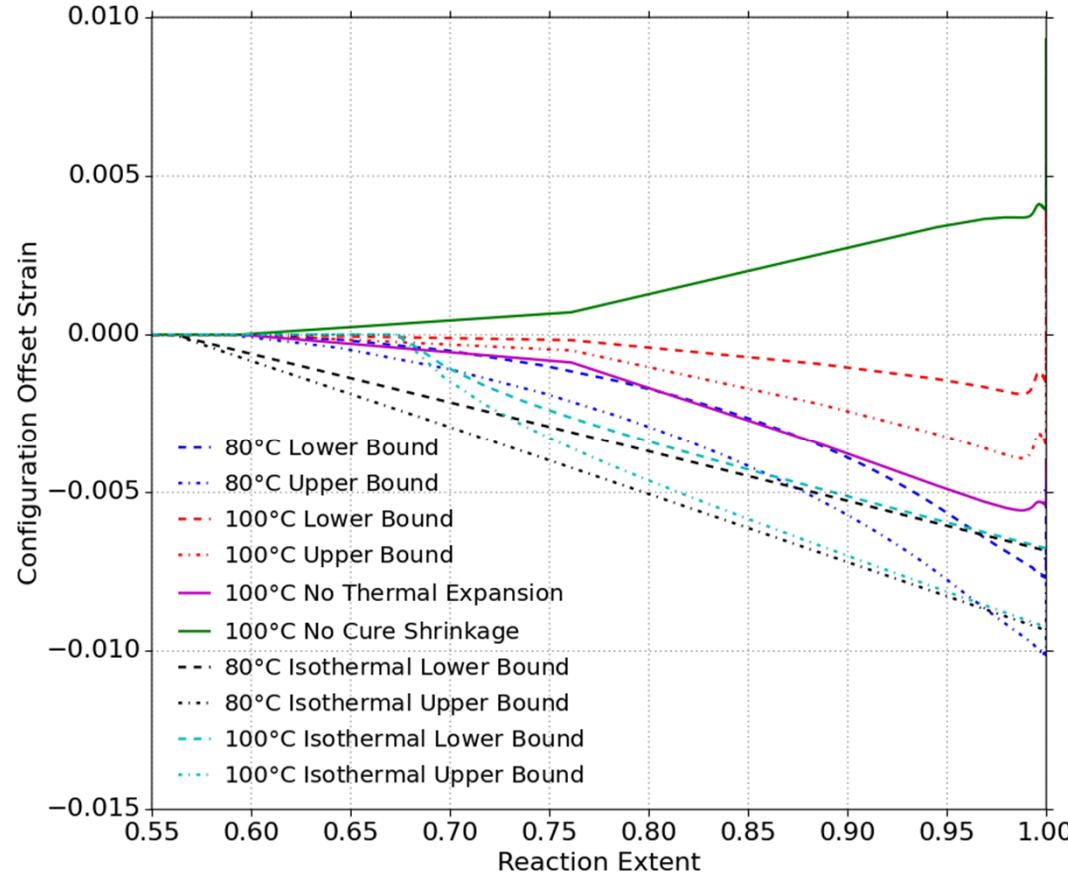


# Solution to the Evolving Configuration Model

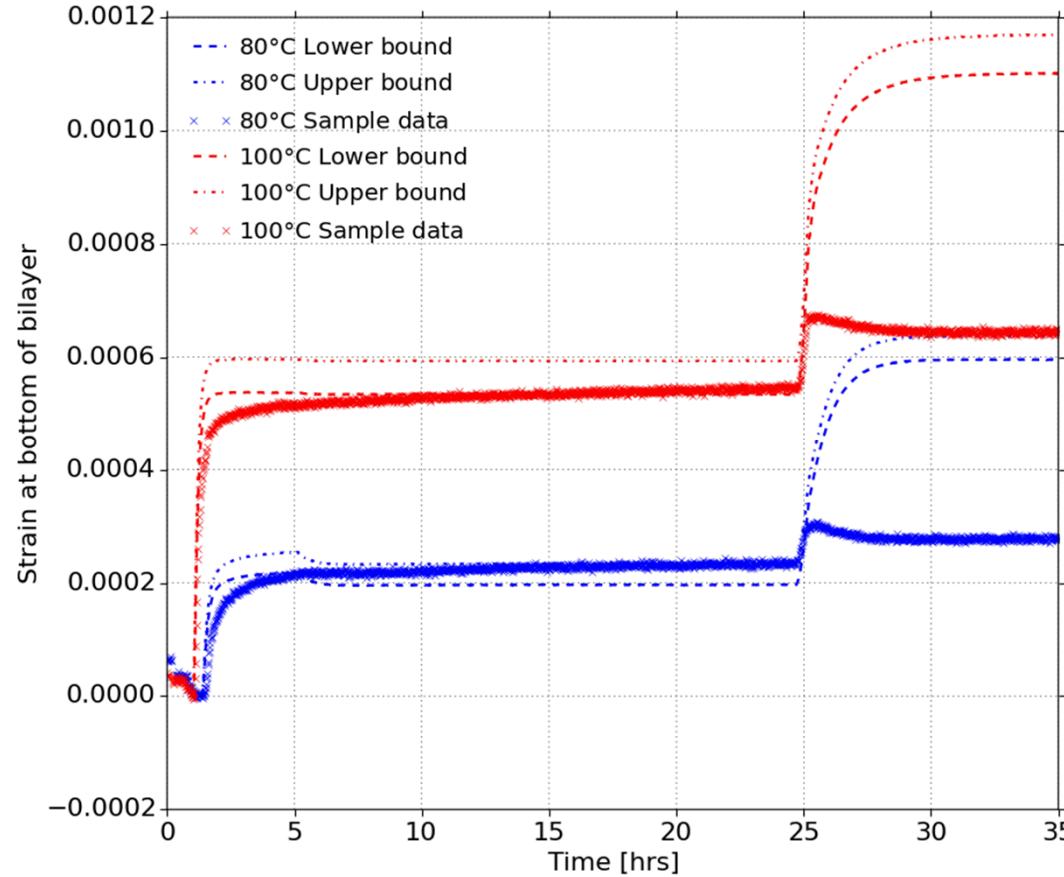
- The evolution of the stress free configuration depends on the history of the deviatoric strain
- Solve for other values
  - Curvature
  - In-plane Strain at boundary
  - Out-of-plane Strain at boundary
  - Assuming in-plane directions are equal, solve for deviatoric in-plane strain

$$\boldsymbol{\varepsilon} = \begin{bmatrix} \varepsilon_{pp} & 0 & 0 \\ 0 & \varepsilon_{pp} & 0 \\ 0 & 0 & \varepsilon_{tt} \end{bmatrix}$$

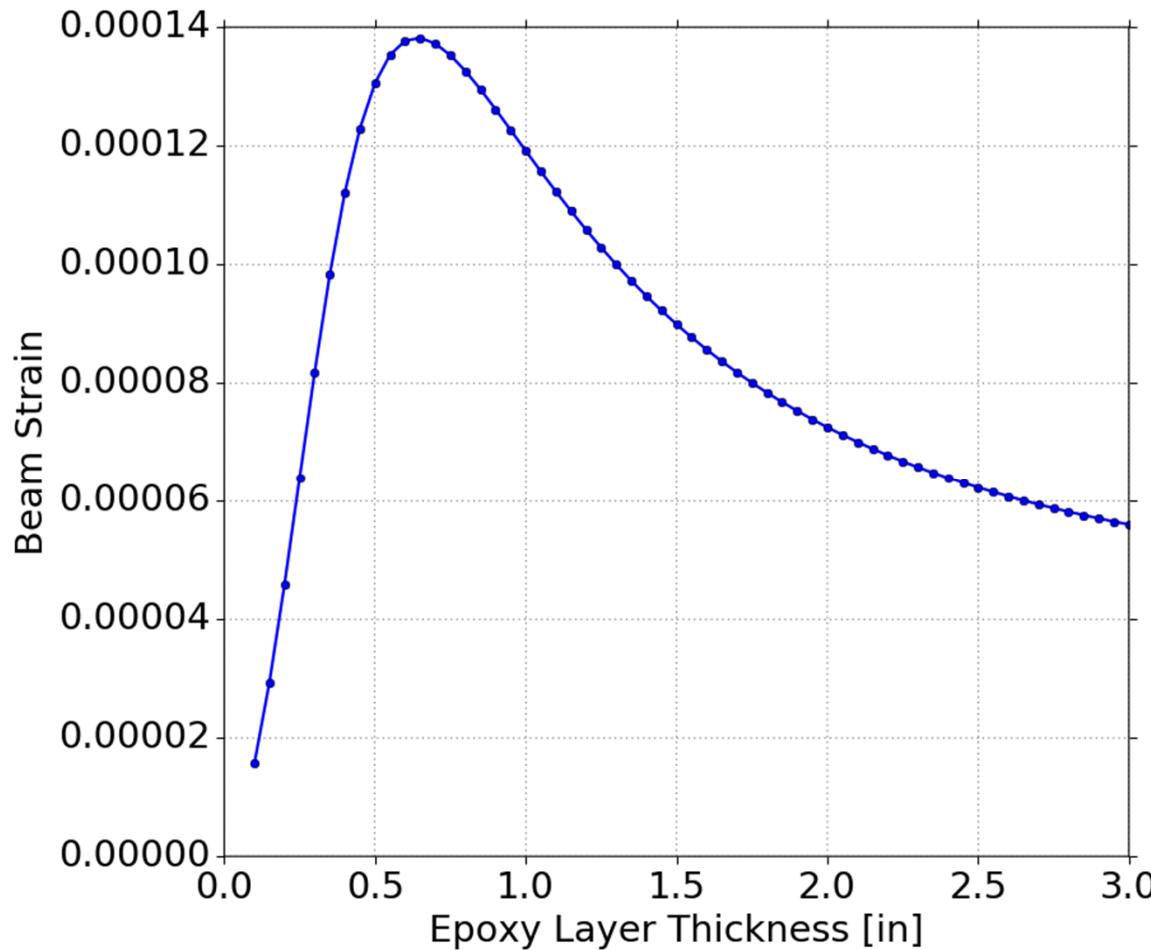
# Model Suggests a Small Change in Stress Free Configuration



# Model Closely Predicts Strain in Experiments

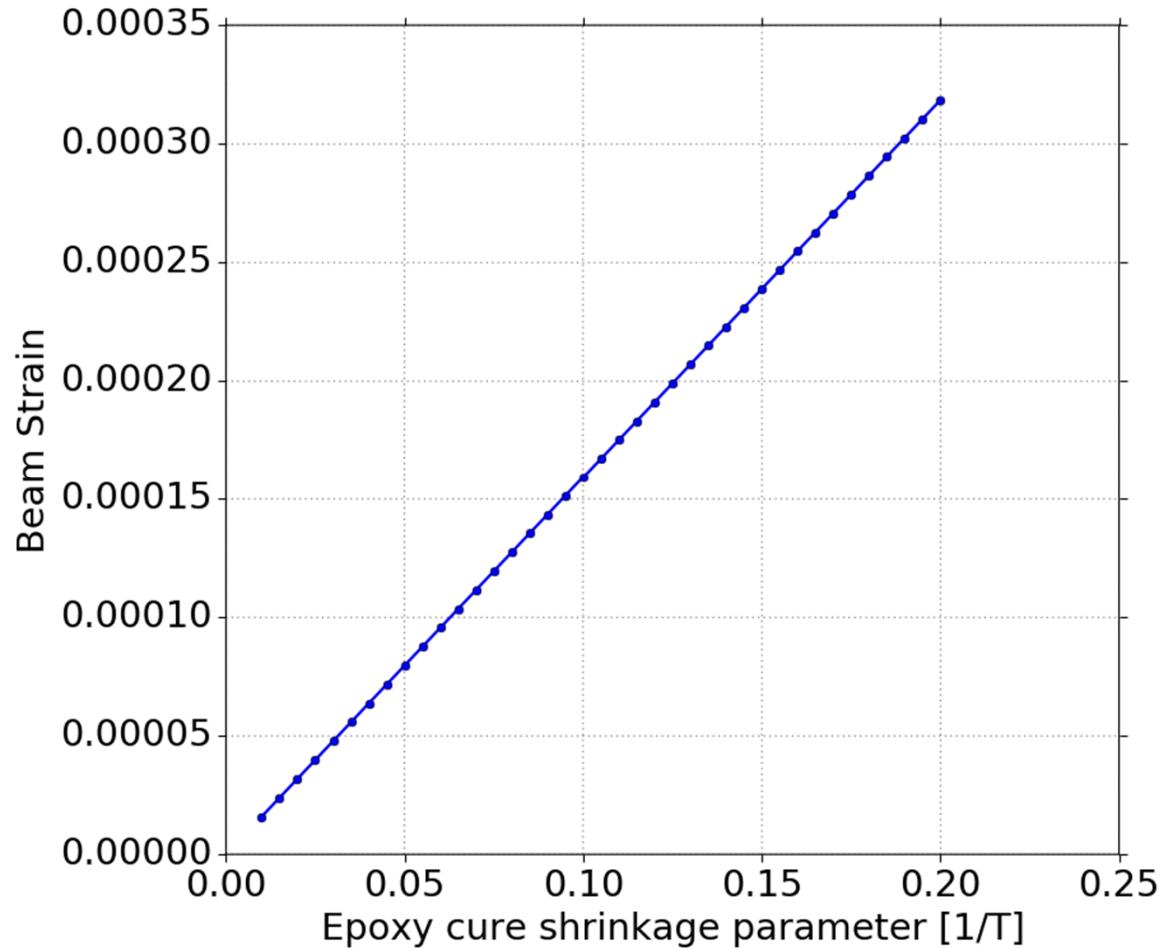


# Sensitivity of Strain Output to Epoxy Thickness

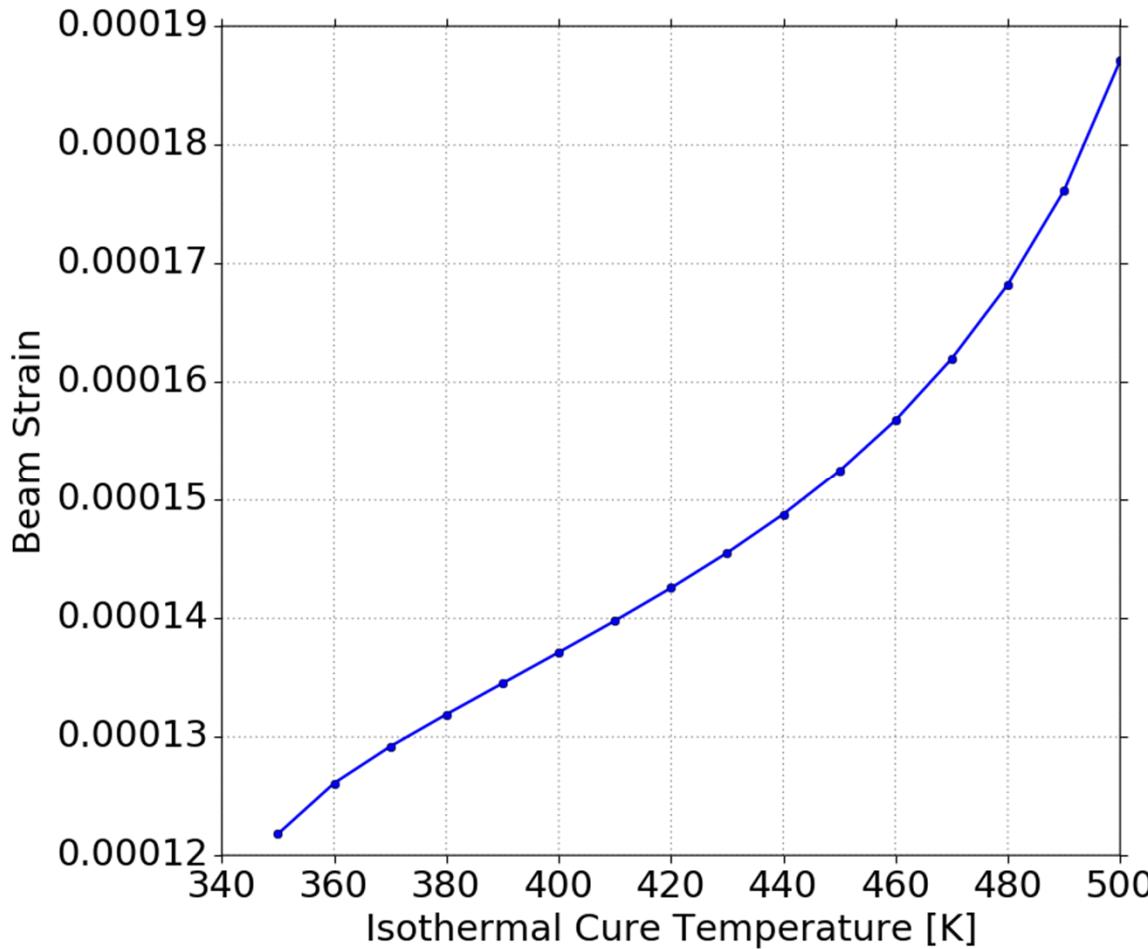


# Sensitivity of Strain Output on Cure Shrinkage Rate

Sensitivity on cure shrinkage rate scales proportionally with beam strain with change in thickness



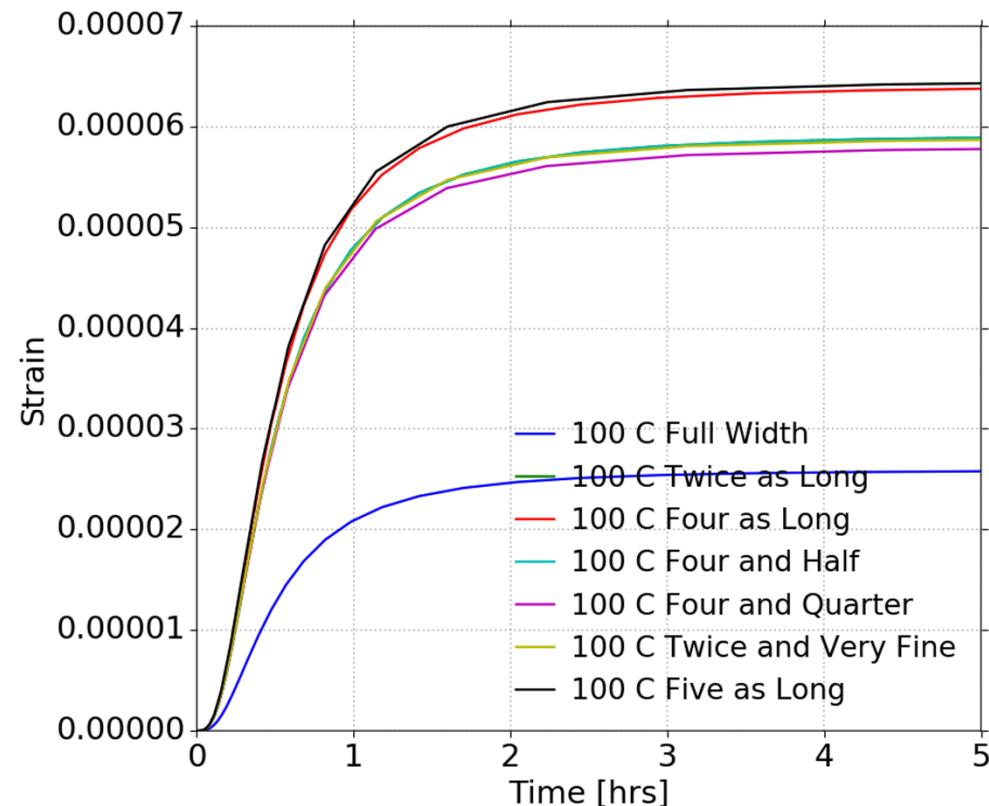
# Sensitivity of Beam Strain on Cure Temperature



# FEA VALIDATION

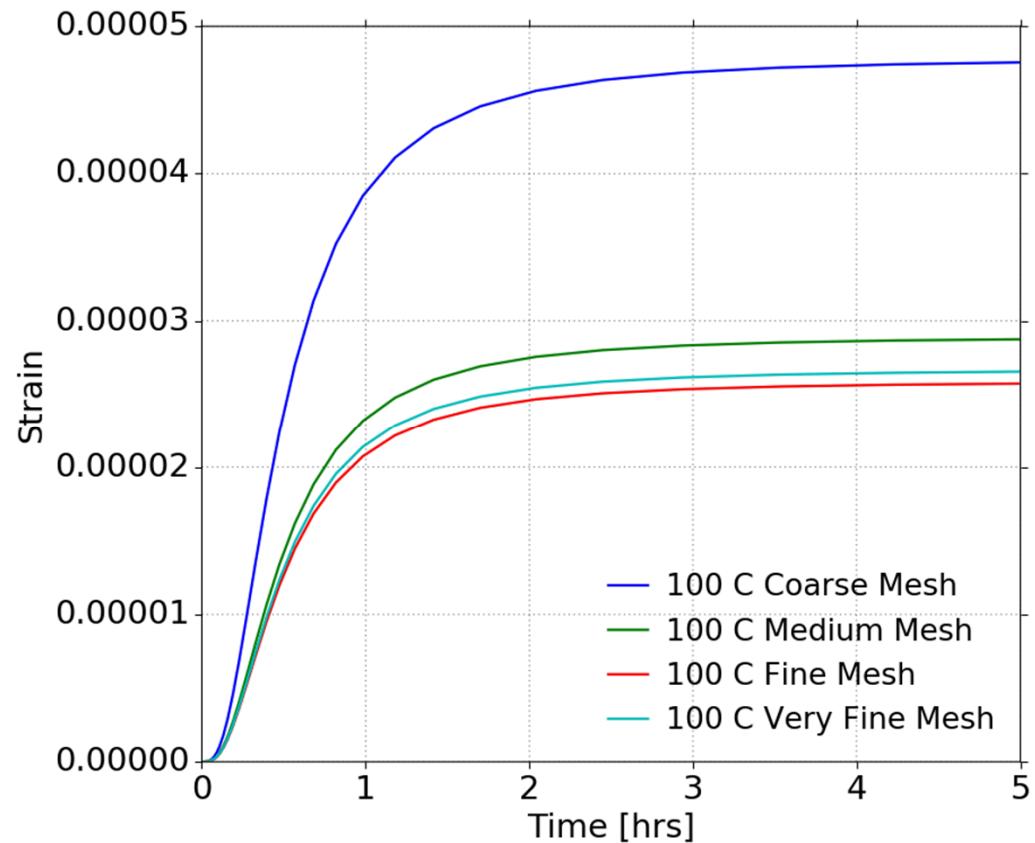
# Finite Element Model Setup

- Geometries of various lengths – 1”, 2”, 4”
- Quarter symmetry exercised
- Final Geometry: 1” x 4” with quarter symmetry (Mesh: 0.5” x 2”)



# FEA Mesh at 3 Densities

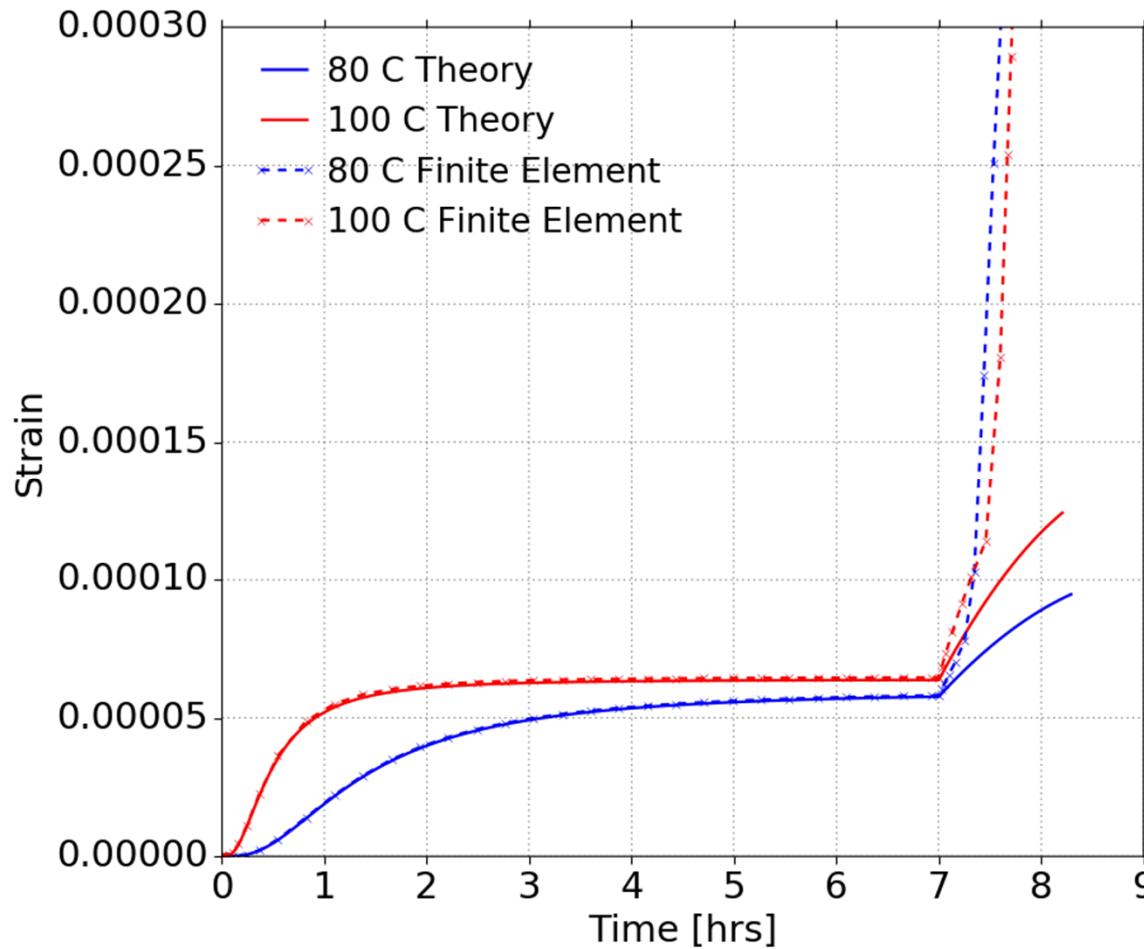
- Coarse, Medium, Fine meshes to prove mesh convergence
  - ~5K elements
  - ~38K elements
  - ~300K elements



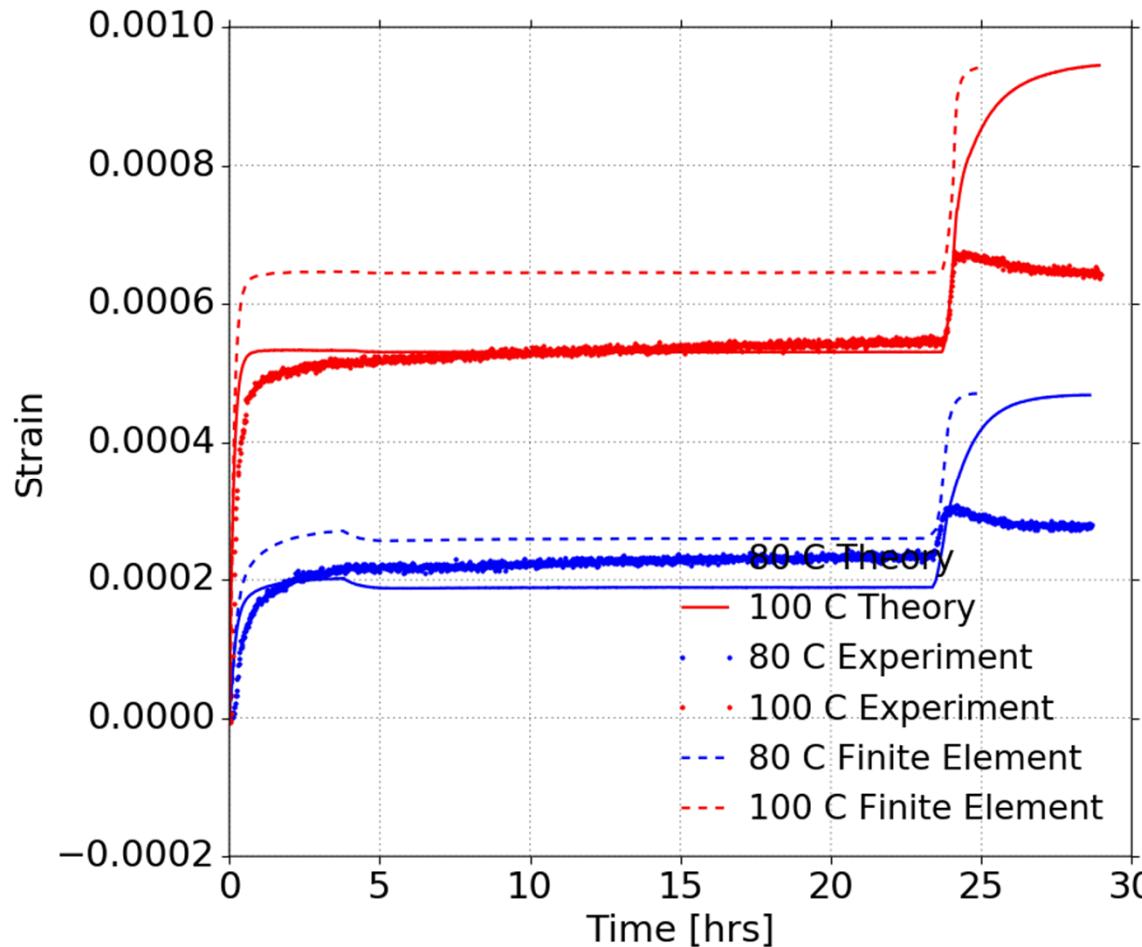
# Material Models

- Linear Elastic Aluminum
  - Small strains, good assumption
- Universal Curing Model

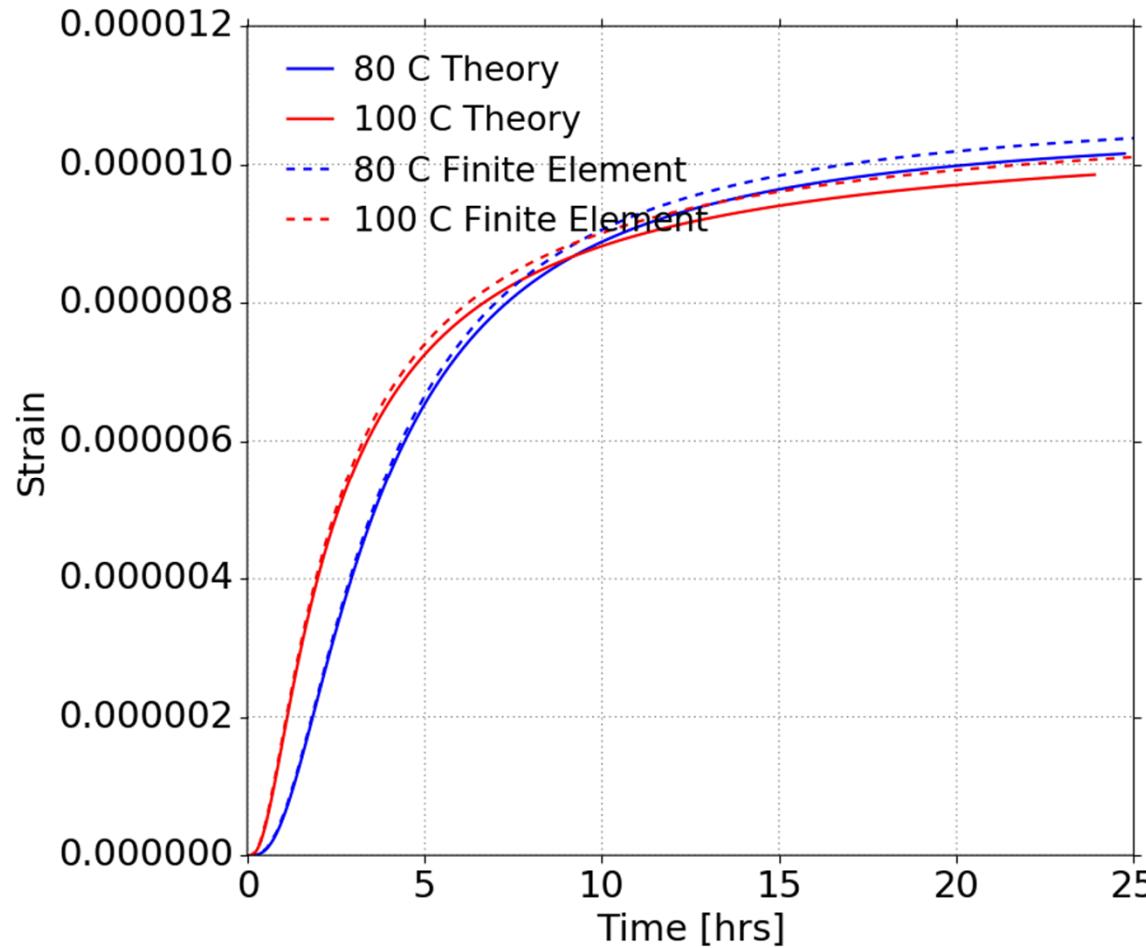
# Comparison of FEA and Theory for Isothermal Case



# FEA, Theory, and Experiment Comparison



# Preliminary DEA Results



# Conclusions

- For 828/T403:
  - Analytical model captures major strain behaviors from the bilayer beam experiment.
  - FEA and Analytical models show a discrepancy for changing temperatures – possibly points to a difference in treatment of thermal strain
- For 828/DEA:
  - FEA and Analytical models agree fairly well for isothermal case
  - Experimental data is inadequate for comparison, still additional understanding needed of the 828/DEA Epoxy