



Veeco D-125 MOCVD system

## Growth and Electrical Properties of AlGaN-based PN Diodes and High-Electron-Mobility Transistors

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### *Acknowledgements:*

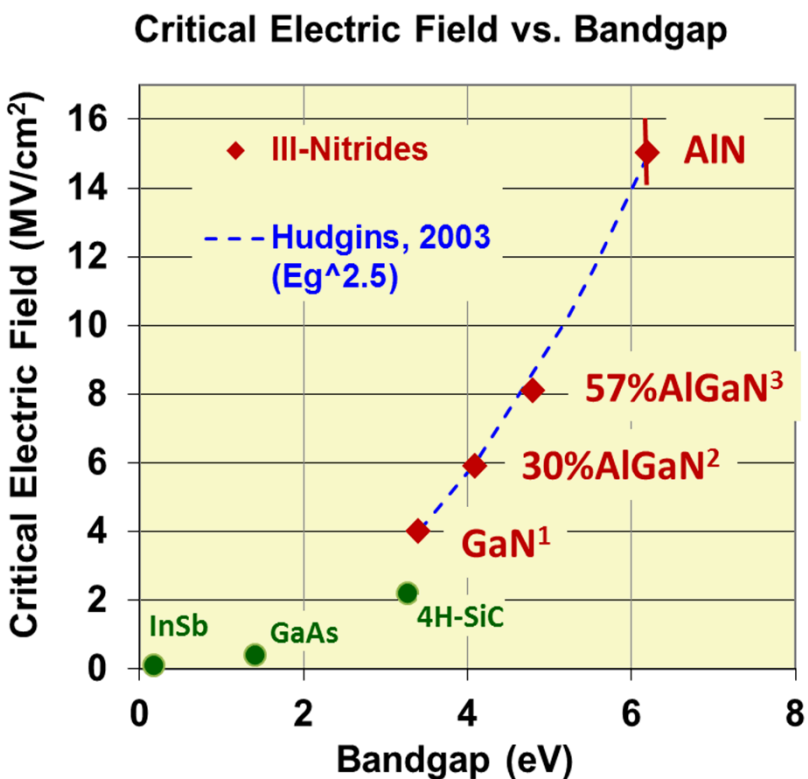
M. Smith, K. Cross, V. Abate and L. Alessi

- Wide bandgap AlGa<sub>N</sub> alloys for power electronics
- PN diodes Al<sub>0.3</sub>Ga<sub>0.7</sub>N
  - Growth of thick, N- drift layers
  - Breakdown voltage and forward ON resistance
- Al-rich Al<sub>x</sub>Ga<sub>1-x</sub>N ( $x > 0.7$ ) heterostructures for HEMTs
  - 2DEG formation (growth conditions, doping & structure)
  - Initial transistor characteristics
- Summary

# Ultra-wide-bandgap semiconductors (UWBS, >4eV) for power electronics

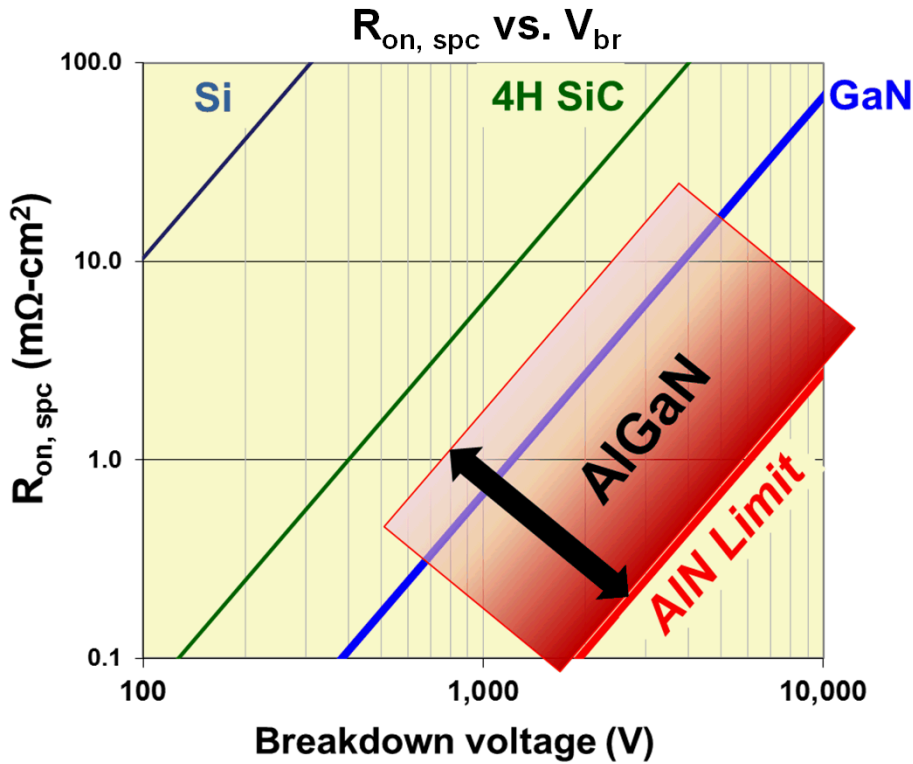
## Critical Electric Field (MV/cm)

$$E_c \propto E_g^{2.5} \quad (\text{Direct gap})$$



## Unipolar Figure of Merit (vertical devices)

$$UFOM = \frac{V_{br}^2}{R_{on,sp}} = \frac{1}{4} \epsilon \mu E_c^3 \propto E_g^{7.5}$$

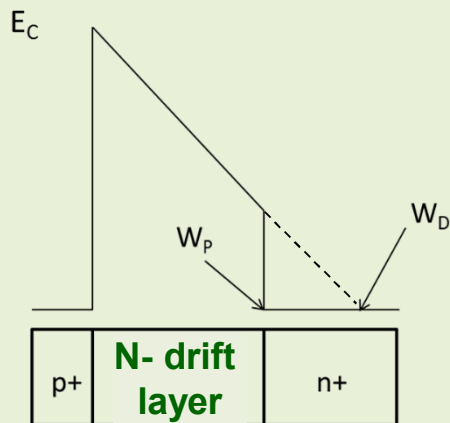


1-Armstrong, EL 2016, 2-Allerman, EL 2016, 3-Nisikawa, JpnJAP 2007

# Breakdown voltage of drift layer: doping and thickness

## Punch-Through

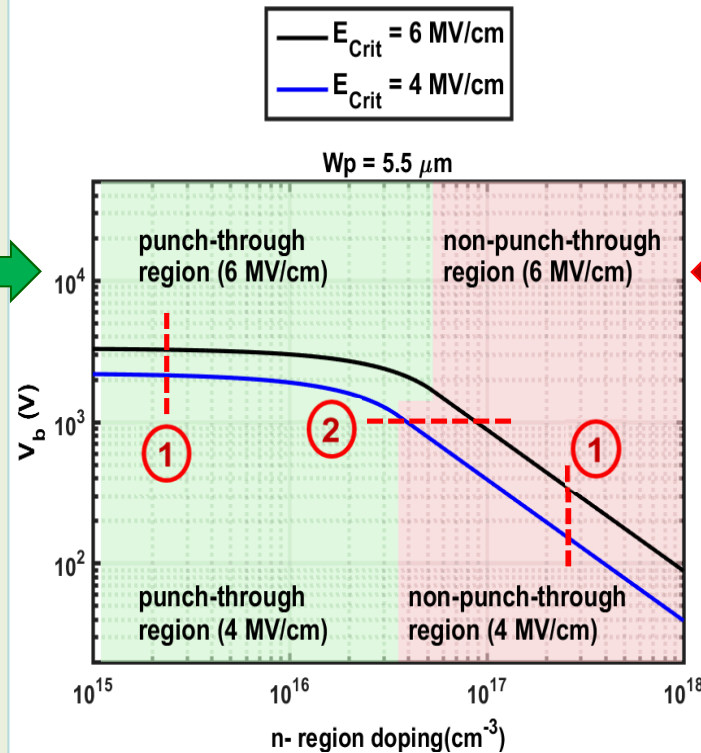
$$W_P < W_D$$



$$V_{br} = E_C W_P - \frac{q N_D W_P^2}{2 \epsilon}$$

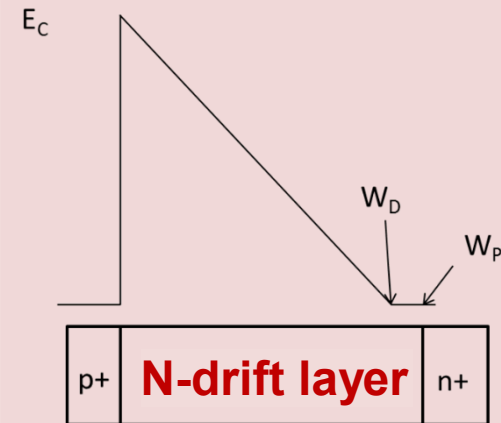
- $N_d$  is lower than optimal

➔ Thickness determines  $V_{br}$



## Non-Punch-Through

$$W_P \geq W_D$$



$$V_{br} = \frac{\epsilon E_C^2}{2 q N_D}$$

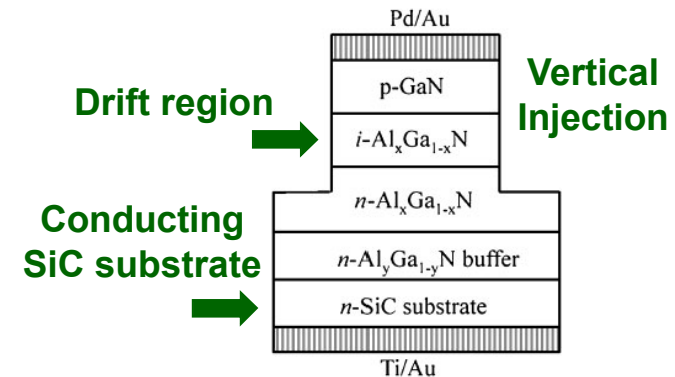
- Drift layer is thicker than optimal

➔  $N_d$  determines  $V_{br}$

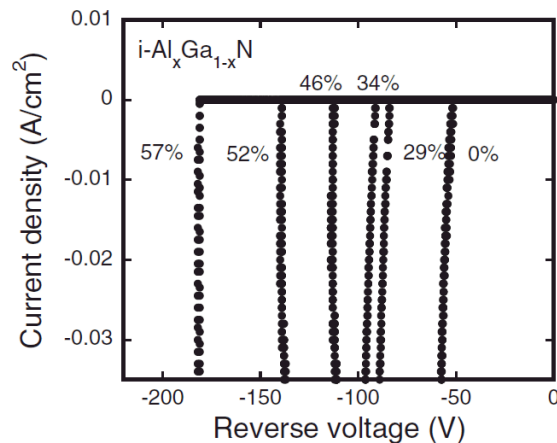
# Prior AlGa<sub>x</sub>N PIN diode results (Nishikawa, NTT, 2007)

## Al<sub>x</sub>Ga<sub>1-x</sub>N Vertical PN diode ( $0 < X_{Al} < 0.57$ )

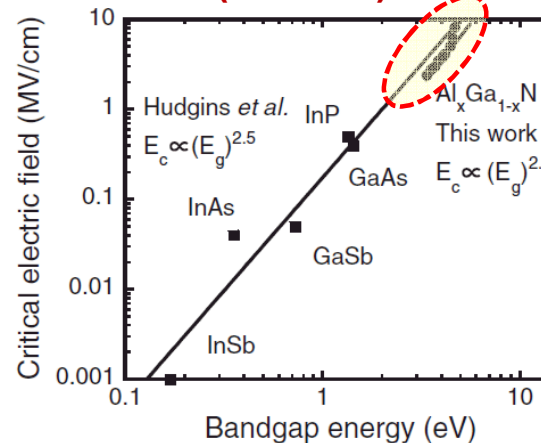
- Drift Layer:  $\sim 0.2 \mu\text{m}$ ,  $N_o \sim 2 \times 10^{16} \text{ cm}^{-3}$
- N-SiC substrates,  $R_{on,sp} = 1.45 \text{ m}\Omega\text{-cm}^2$  ( $X_{Al}=0.22$ )



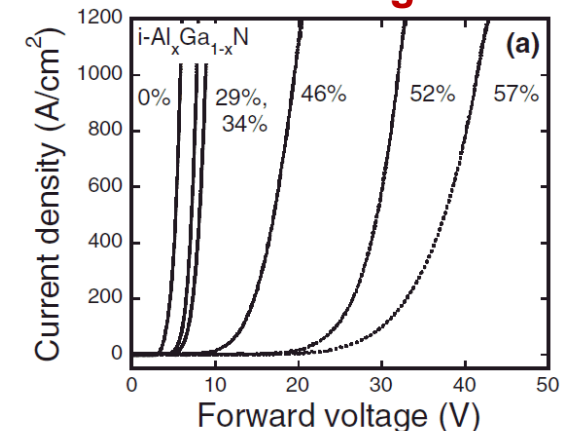
### Reverse Breakdown < 200V



### $E_{crit} \sim 8 \text{ MV/cm}$ (2x GaN)



### Higher forward turn-on for increasing Al



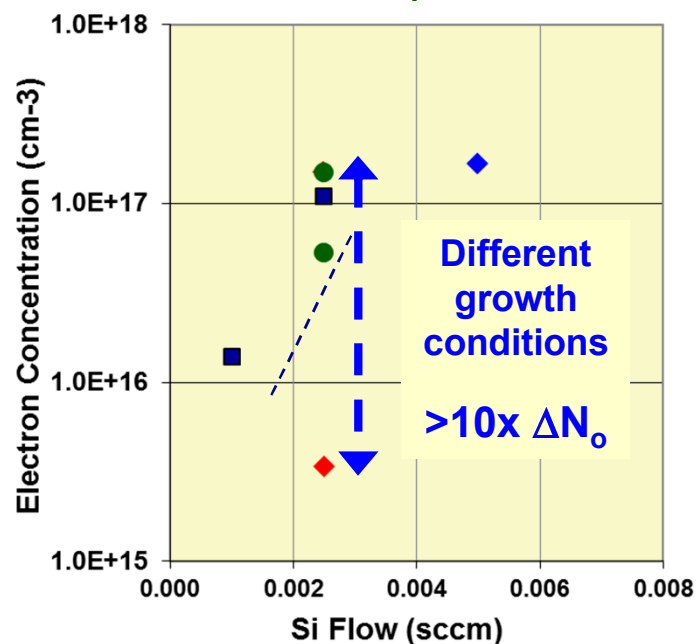
- ➔ Breakdown voltage increases with larger bandgap
- ➔ Critical electric field scales as  $E_g^{2.7}$



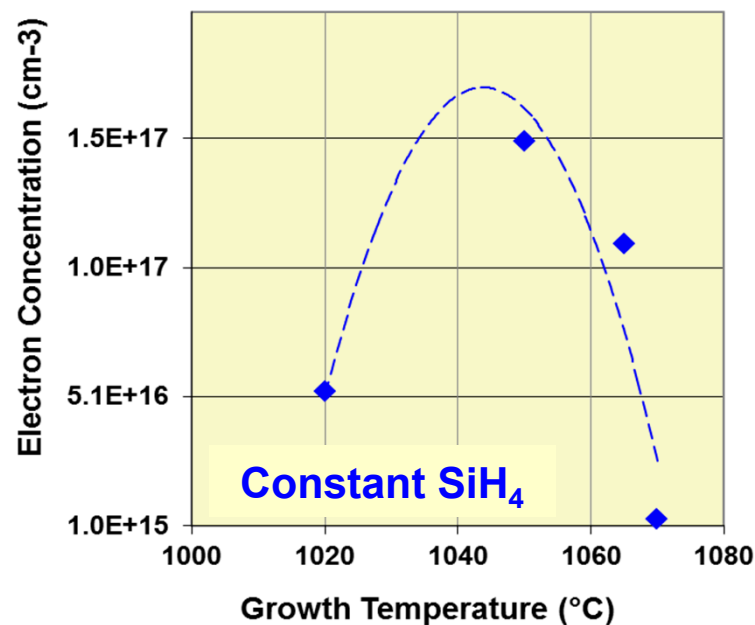
# Control of N-type doping of drift region ( $\text{Al}_{0.3}\text{Ga}_{0.7}\text{N}$ )

➔ *Si incorporation is linear in  $\text{SiH}_4$  flow but...*

*Electron concentration (hall)  
vs.  $\text{SiH}_4$  flow (30%AlGaN)*



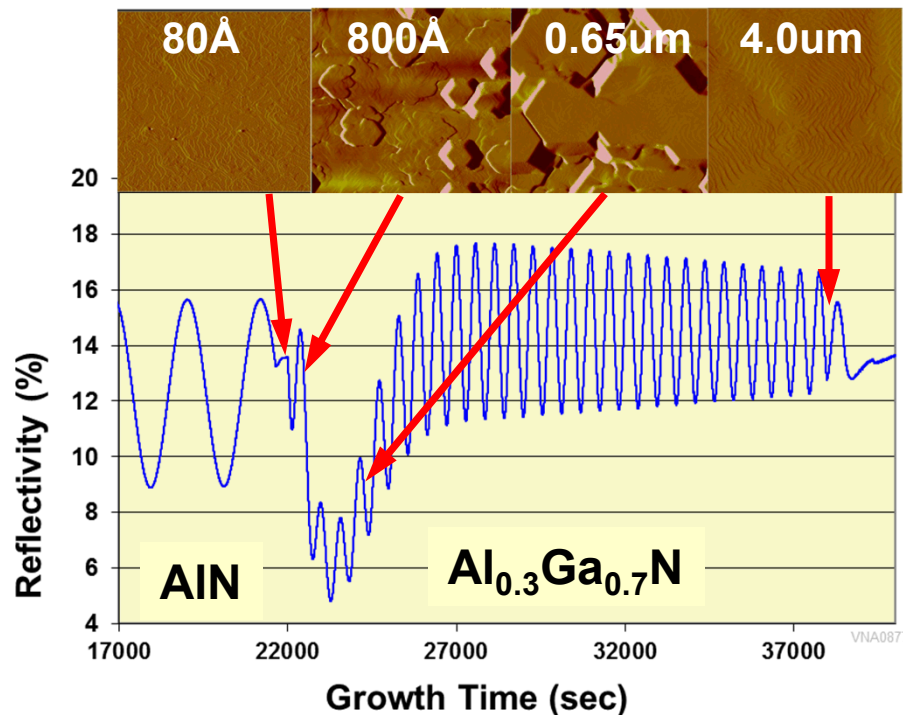
*Electron concentration (hall)  
vs. growth temperature (30%AlGaN)*



➔ *... electron concentration is highly dependent on  
growth condition (i.e. the density of compensating defects)*

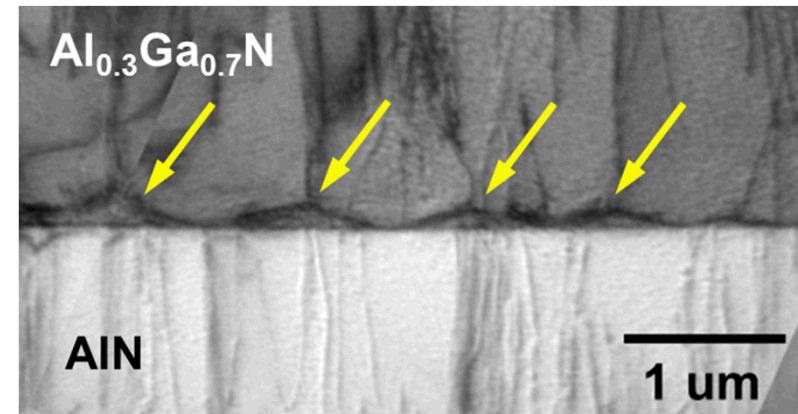
# Relaxation and TDD reduction in $\text{Al}_{0.3}\text{Ga}_{0.7}\text{N}$ on $\text{AlN}$

## In-situ Reflectance and AFM



- Surface roughens (3D growth) and then planarization (2D growth)
- $\text{Al}_{0.3}\text{Ga}_{0.7}\text{N}$  layer is >80% relaxed, TDD  $\sim 2\text{-}4 \times 10^9 \text{ cm}^{-3}$

## Cross-section TEM



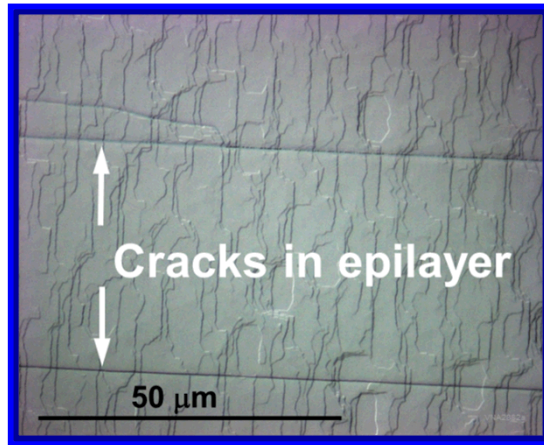
S. Lee, D. Follstaedt, J. Floro 2009

- Strain is relieved by interfacial misfit dislocation networks with hexagonal boundaries
- Threading dislocations “bundle” into triangular features

➔ *2D to 3D to 2D growth relieves strain & lowers TDD*

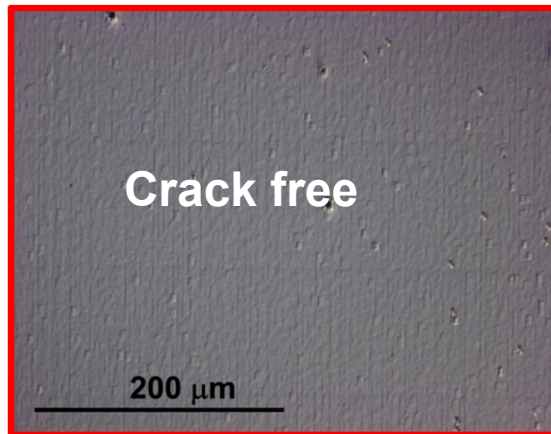
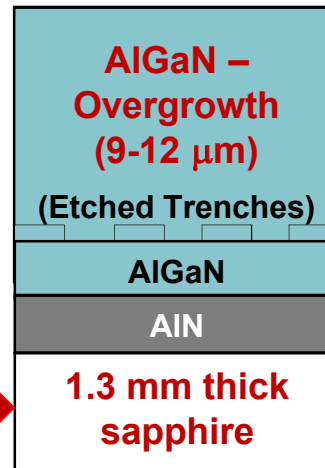
# Reduce cracking and bow by using 3x thicker sapphire

## Optical Image of $\text{Al}_{0.3}\text{Ga}_{0.7}\text{N}$ surface

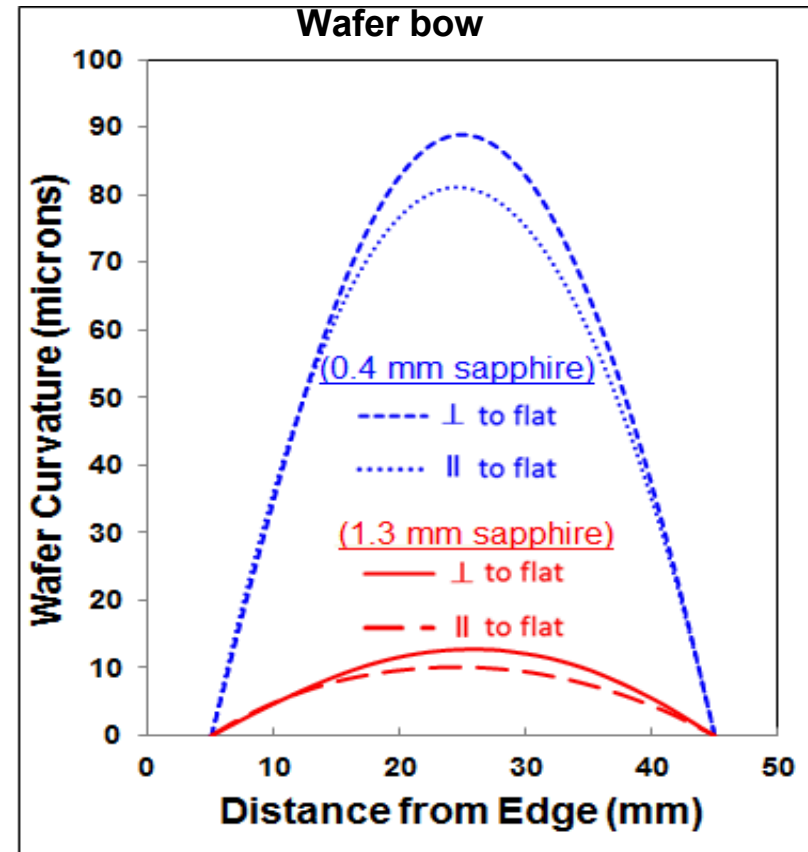


0.4 mm thick sapphire

**Cracking  
& bowing**



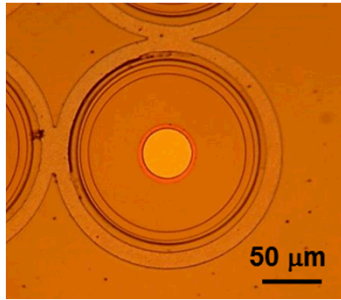
1.3 mm thick sapphire



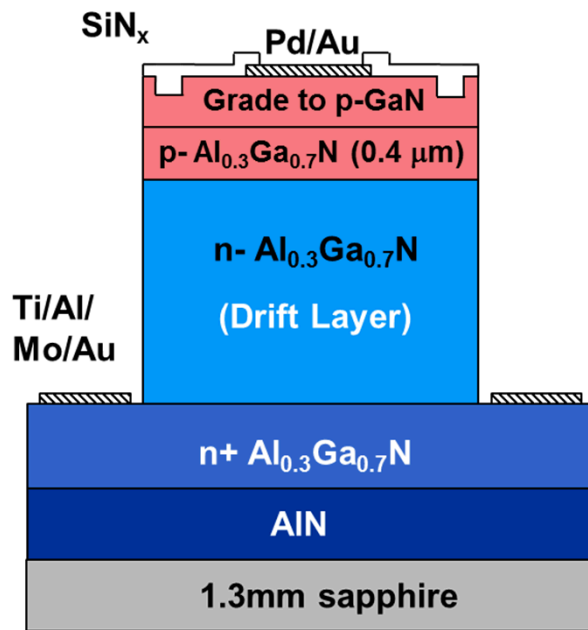
**➡ 3x thicker sapphire reduces cracking & wafer bow**



# $\text{Al}_{0.3}\text{Ga}_{0.7}\text{N}$ “Quasi-Vertical” PN diode on sapphire



Optical image of diode

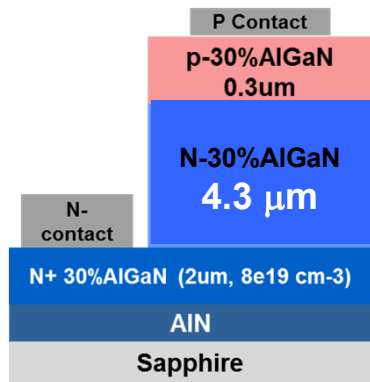


Quasi-Vertical PN diode structure

## $\text{Al}_{0.3}\text{Ga}_{0.7}\text{N}$ PN diode design & fabrication

- Implanted junction edge termination
- Mesa / p-metal: 150 / 50  $\mu\text{m}$
- *Drift Layer: 4.3, 5.5, 7.5, 9, 11 and 15.5  $\mu\text{m}$*
- Crack-free (except 15.5  $\mu\text{m}$  drift layer)
- Total epi thickness: 7 – 18, 22  $\mu\text{m}$
- Drift layer doping: mid  $10^{15}$  – mid  $10^{16} \text{ cm}^{-3}$   
Mobility: 150  $\text{cm}^2/\text{Vs}$
- Sheet resistance: 40-80  $\Omega/\text{sqr.}$
- *Threading dislocation density:  $1\text{-}3 \times 10^9 \text{ cm}^{-2}$*

# Al<sub>0.3</sub>Ga<sub>0.7</sub>N “Quasi-Vertical” PN diode on sapphire



- Drift layer: 4.3 μm, TDD ~ 1-2e9 cm<sup>-2</sup>

$$N_o \sim 5-7e16 \text{ cm}^{-2}, \mu \sim 150 \text{ cm}^2/\text{Vs},$$

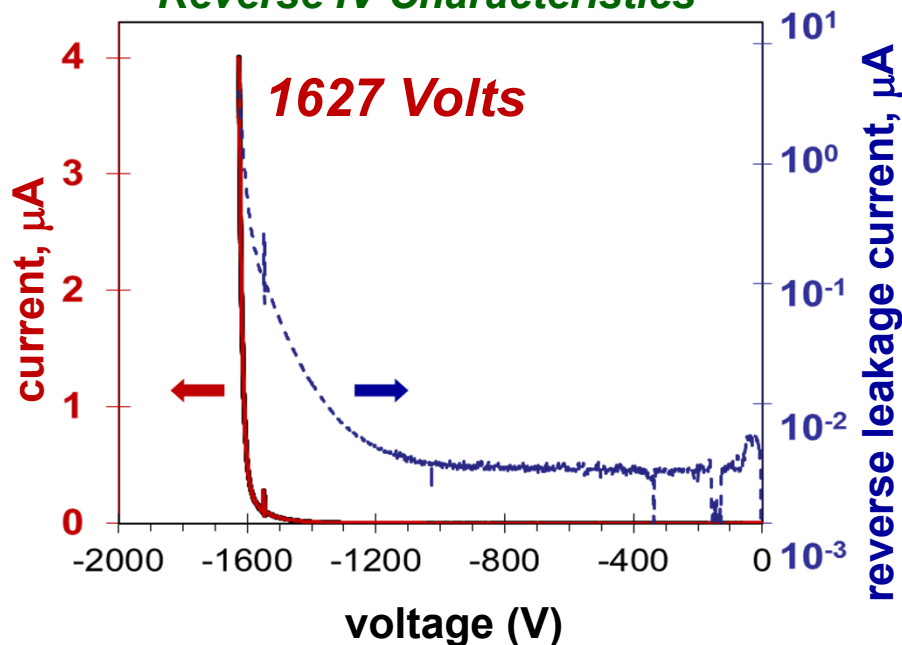
$$\Rightarrow V_{br} > 1600 \text{ Volts}$$

$$\Rightarrow I_{\text{reverse}} < 3 \text{ nA (to 1200V)}$$

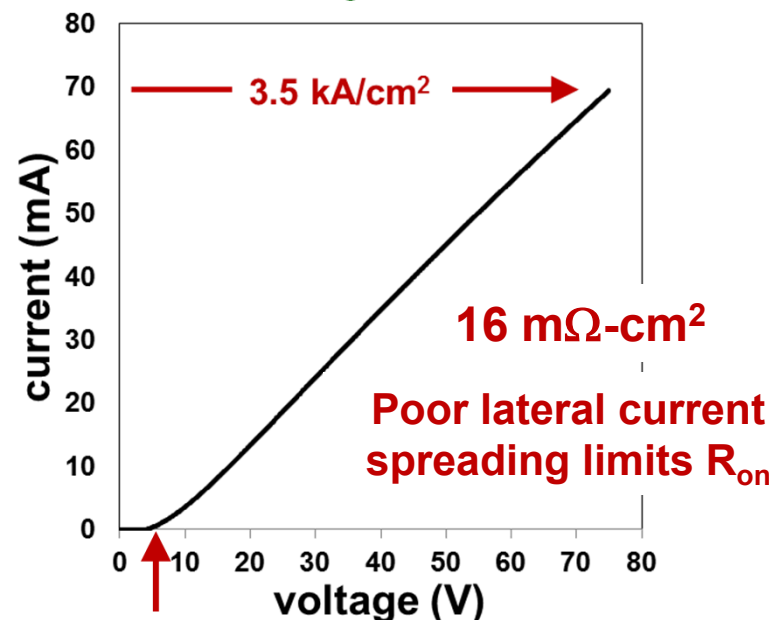
$$\Rightarrow E_C \sim 5.9 \text{ MV/cm}$$

$$\Rightarrow V_{br}^2/R_{\text{on,sp}} = 150 \text{ MW/cm}^2$$

Reverse IV Characteristics



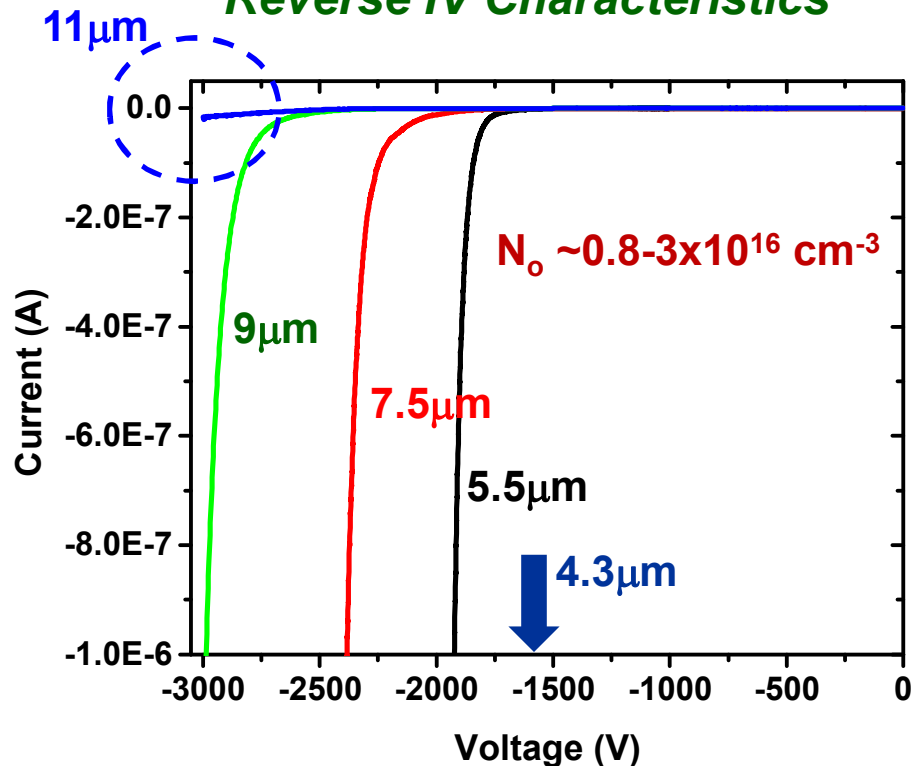
Forward IV Characteristics



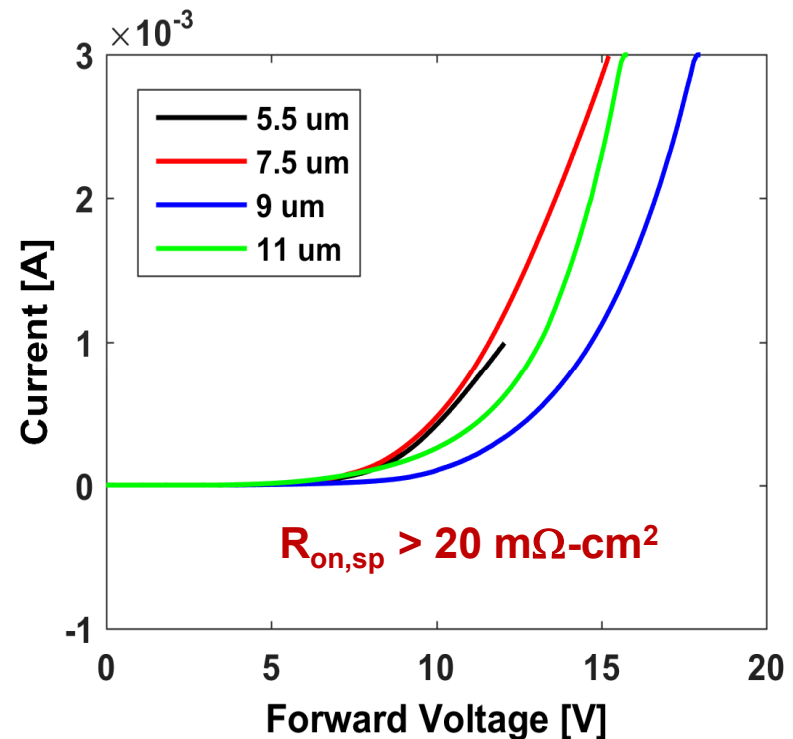
Allerman, Elec. Lett. 2016

# Al<sub>0.3</sub>Ga<sub>0.7</sub>N “Quasi-Vertical” PN diode – thicker drift layers

## Reverse IV Characteristics



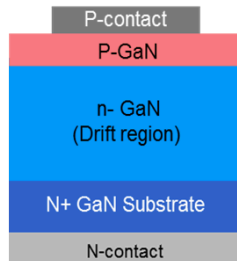
## Forward IV Characteristics



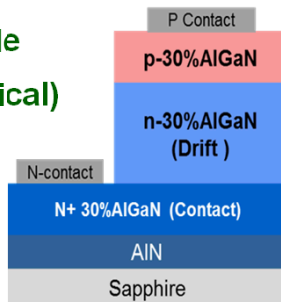
- ➔ Breakdown voltages  $\sim 3000\text{V}$  (Drift: 9 & 11  $\mu\text{m}$ )
- ➔  $V_{\text{br}}$  increases with drift layer thickness for  $N_o$

# Breakdown voltages reported for III-Nitride PN diodes

**GaN diode  
(vertical)**



**AlGaN diode  
(Quasi-vertical)**



## III-N PN diodes with > 3 kV breakdown voltage

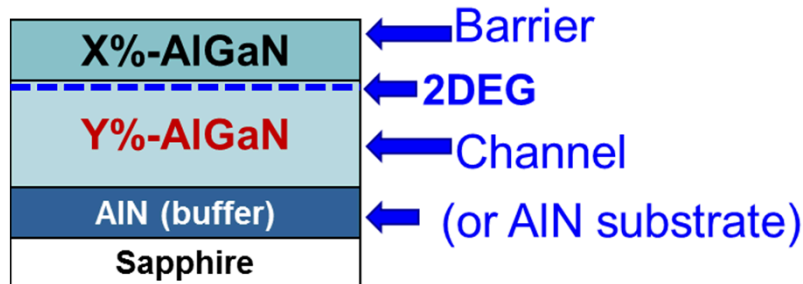
Breakdown (kV)	No (cm <sup>-3</sup> )	Drift (μm)	Material	Group	Ref
4.7	2/9/16e15	33	GaN	Hosei Univ.	EDL 36p1180_2015
4.0	2-5e15	40	GaN	Avogy	EDL 36p1073_2015
3.9	3e15	30	GaN	Sandia	EL 52p1170_2016
3.7	5e15	>30	GaN	Avogy	EDL 35p247_2014
3.48	1/3/12e15	32	GaN	Hosei Univ.	IEDM15-237_2015
>3	0.8-3e16	11	30%-AGaN	Sandia	This work
3.0	0.8-3e16	9	30%-AGaN	Sandia	This work
3.0	1/10e15	20	GaN	Hitachi	Jpn J Appl Phys 52 p028007_2013

## Advantages of wide-bandgap III-Nitride

	GaN	Al <sub>0.3</sub> Ga <sub>0.7</sub> N	
N <sub>o</sub> (cm <sup>-3</sup> )	low e15	low e16	] ← Larger E <sub>c</sub> (larger E <sub>g</sub> )
Drift (μm)	20-30	~10	
TDD (cm <sup>-2</sup> )	≤ 1e6	low 1e9	← ??



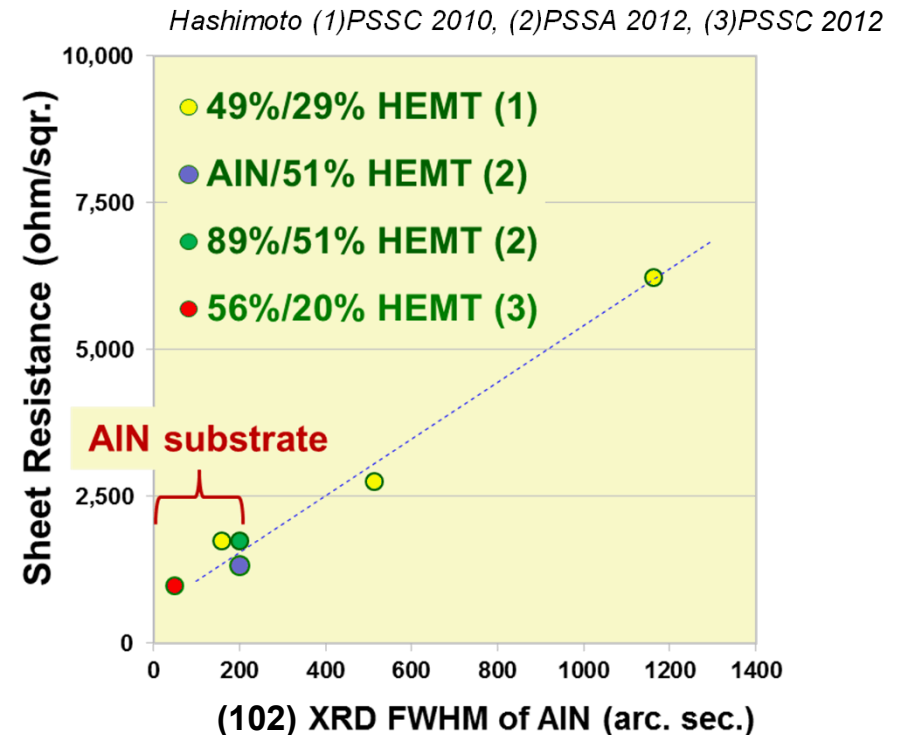
## Typical heterostructure



## Al-rich AlGaN Heterostructures

- $N_s$ :  $0.2 - 2.5 \times 10^{13} \text{ cm}^{-2}$
- Mobility: 140-170 (259,  $X_{\text{Al}}=0.2$ )  $\text{cm}^2/\text{Vs}$
- No 2DEG reported for channel  $X_{\text{Al}} > 0.6$
- Ohmic contacts are difficult
- Low threading dislocation density (TDD) is critical
- *AlN Substrates are expensive & small*

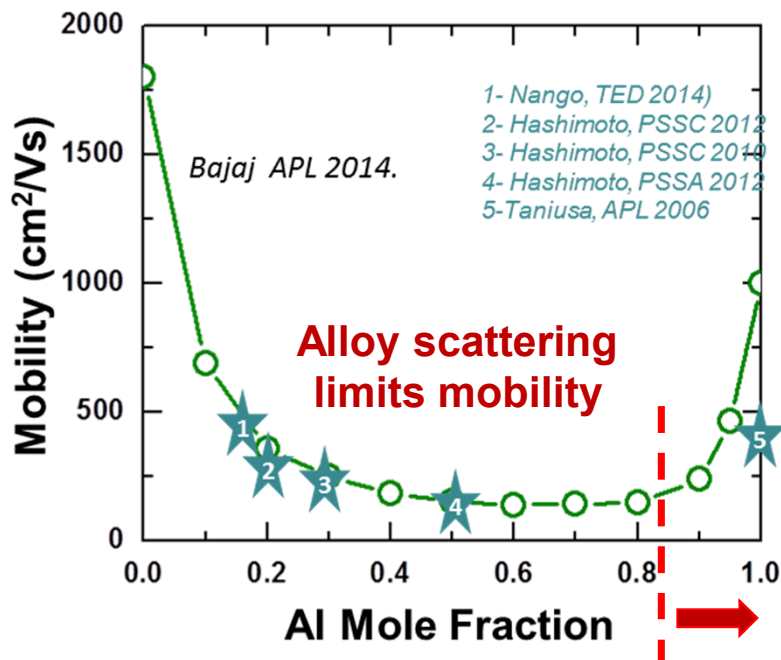
## Sheet Resistance vs. Dislocation Density



**Q: Is there an alternative to AlN substrates?**

# Why higher Al compositions for AlGaN heterostructures?

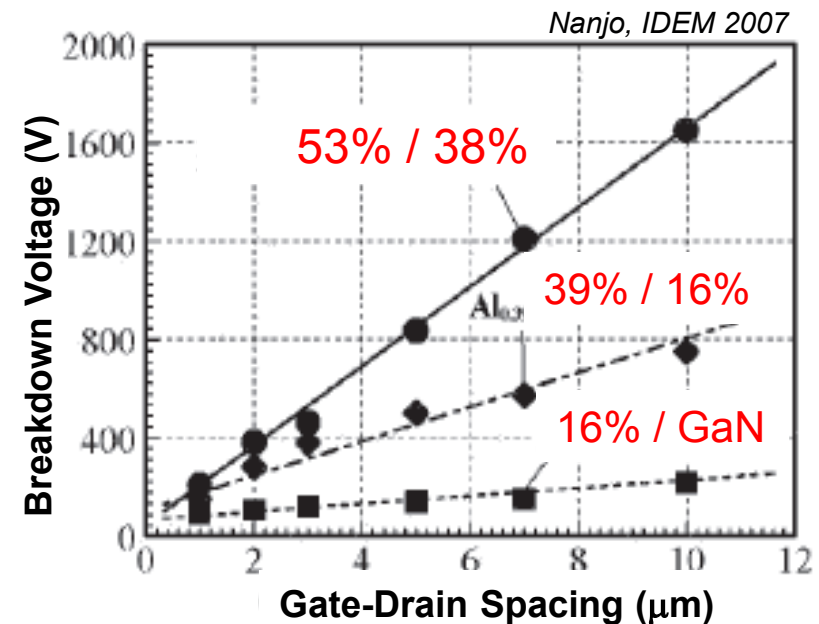
## Calculated Electron Mobility vs. AlGaN Channel Composition



Higher Al compositions:

➔ *higher mobility is predicted*

## Breakdown voltage of AlGaN HEMTs vs. G-D spacing

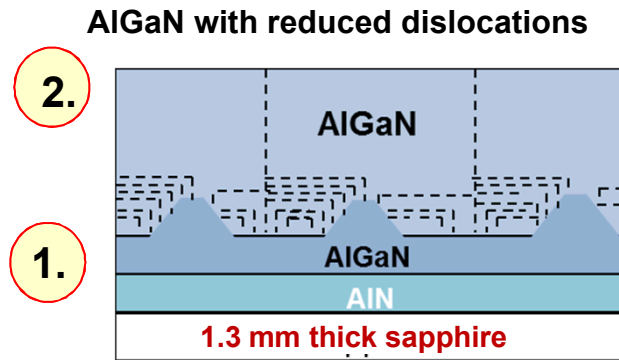


Higher Al compositions:

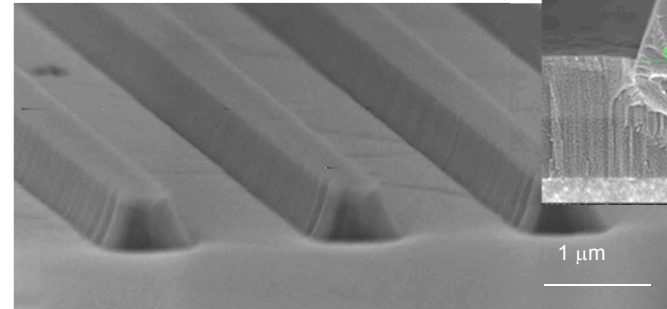
➔ *higher breakdown voltages*

# SNL AlGaN overgrowth of patterned AlGaN templates

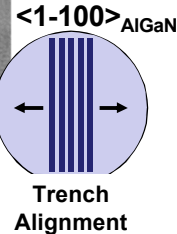
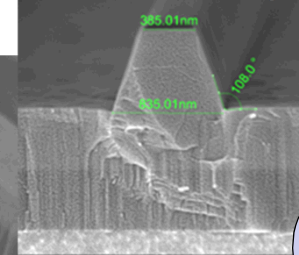
## AlGaN Growth on Patterned Templates



Trenches formed by etching



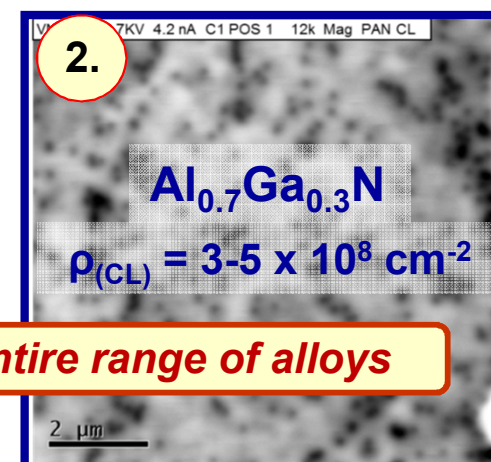
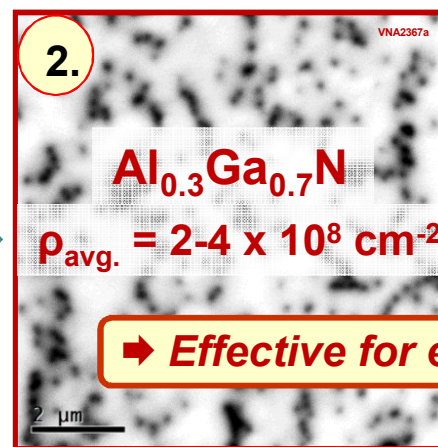
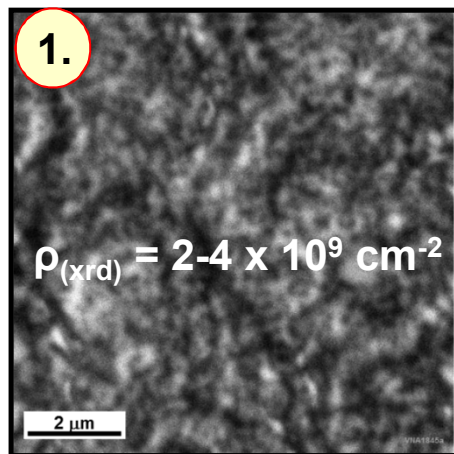
Mesa is 385nm at top!



➔ Sub-micron features are key for uniform reduction of dislocations

Allerman et. al., JCG 2014

## Cathodoluminescence



➔ Effective for entire range of alloys

10-20X reduction

10-15x reduction

# SNL AlGaN overgrowth of patterned AlGaN templates

## AlN overgrowth process (Allerman, JCG 2014)

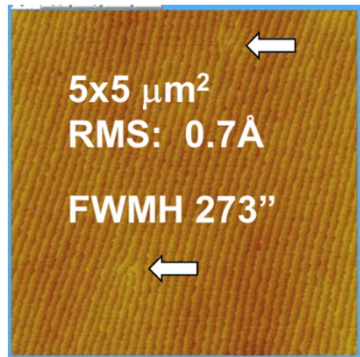


### Etched Pattern

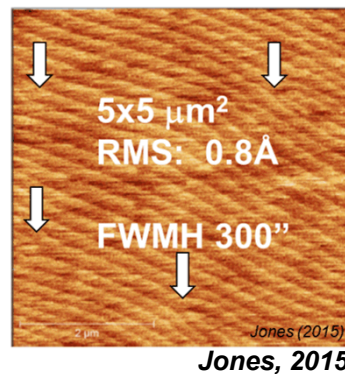
Etched  
Trenches

- 1x1 (mesa, trench, μm)
- 0.2 – 0.7 μm etch depth
- Overgrowth @ 1100°C

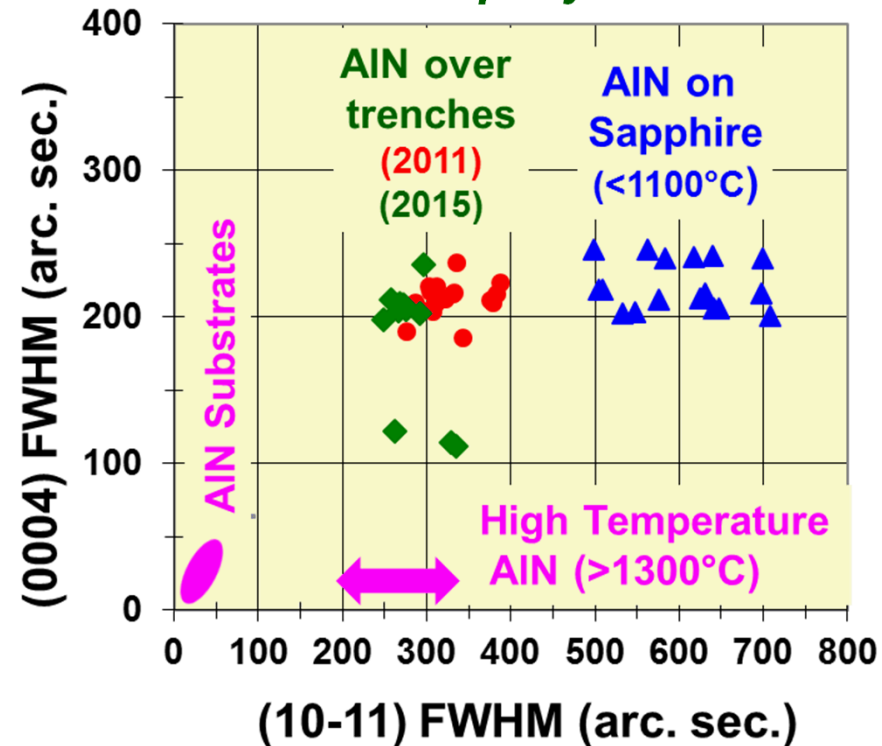
### SNL overgrowth 1100°C



### High Temp. AlN



## X-ray diffraction peak – width of AlN epilayers



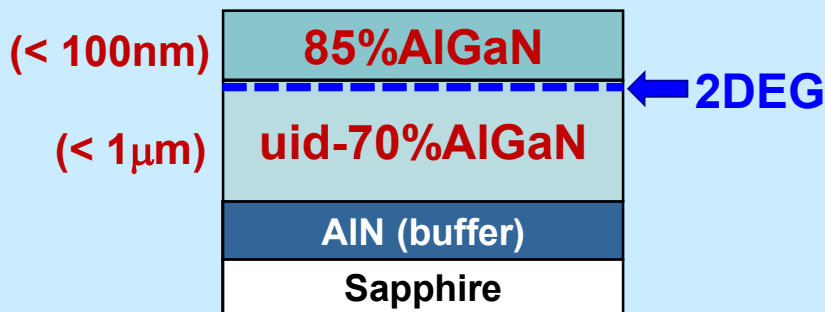
➔ SNL overgrowth @ 1100°C produces AlN similar to HT-AlN

- AlN overgrowth: TDD ~3-5x10<sup>8</sup> cm<sup>-2</sup>



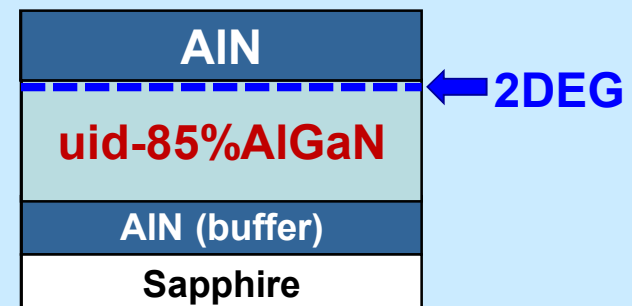
# Al-rich AlGa<sub>0.15</sub>N HEMT heterostructures

## 85% / 70% AlGa<sub>0.15</sub>N HEMT



- Al<sub>0.85</sub>Ga<sub>0.15</sub>N dopes N-type
- Expect lower mobility with lower Al compositions
- Structure could be grown pseudomorphic to AlN substrates

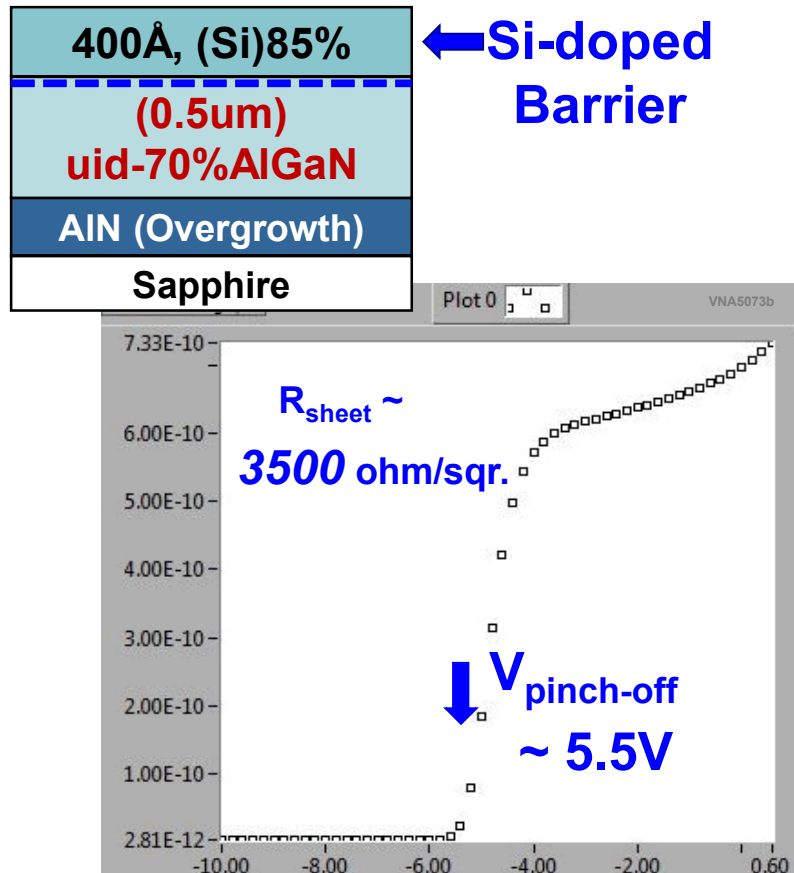
## AlN / 85% AlGa<sub>0.15</sub>N HEMT



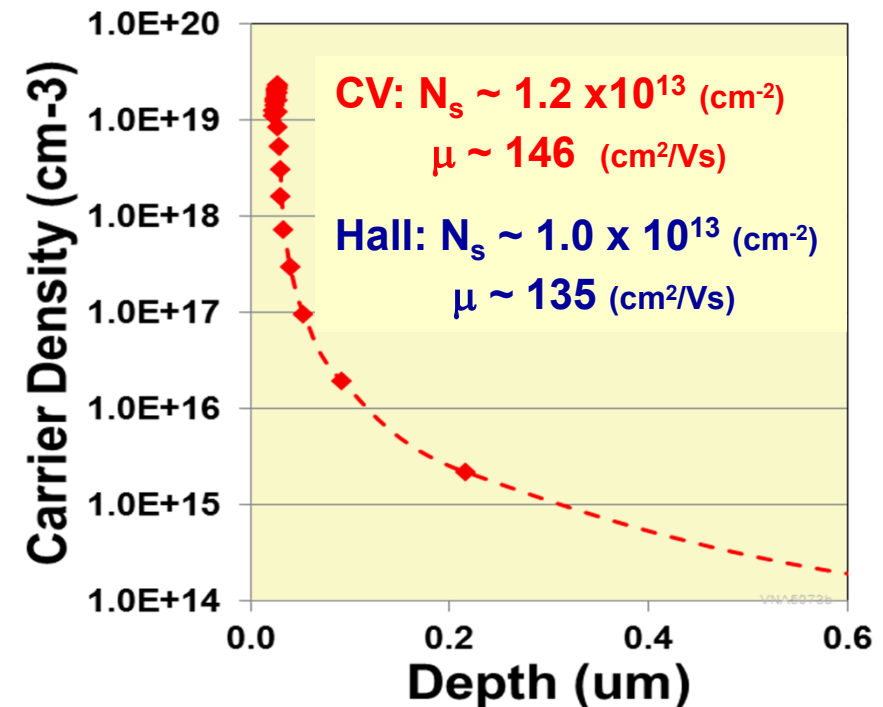
- AlN is hard to dope N-type
- Expect higher mobility with higher Al composition
- Well matched for pseudomorphic growth on AlN substrates

# CV of 85% / 70% AlGa<sub>N</sub> heterostructure

## CV of 85% / 70% AlGa<sub>N</sub> MODFET



## Carrier Density Profile (CV)

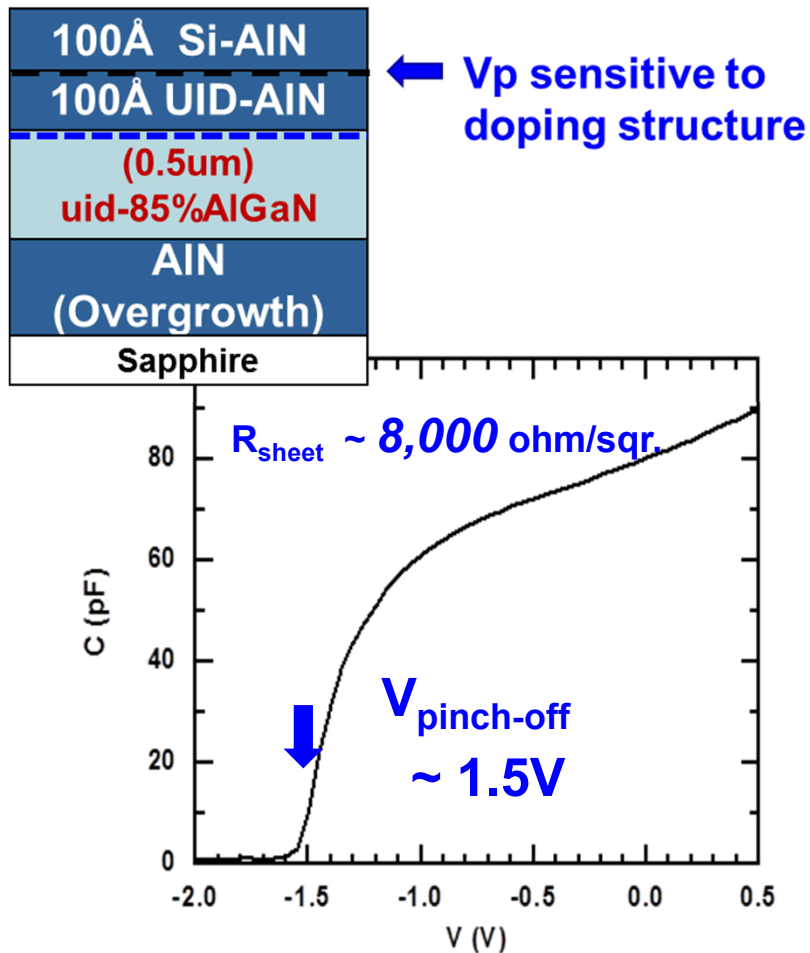


- $N_s$  &  $\mu$  are similar to HEMTs with lower Al channels

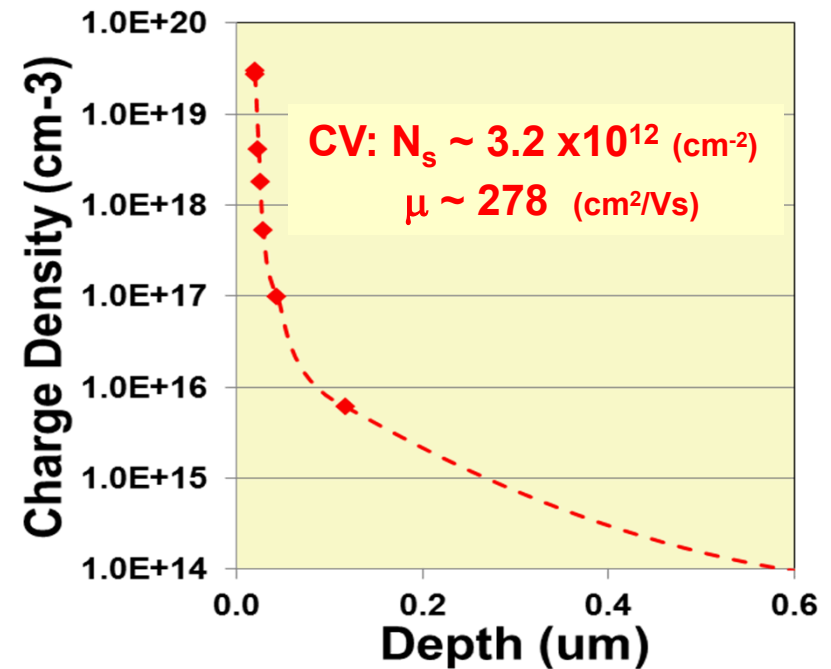
➡ Demonstration of 2DEG in Al<sub>x</sub>Ga<sub>1-x</sub>N channel for  $x > 0.6$

# CV of AlN / 85% AlGaN heterostructure

## CV of AlN / 85%AlGaN MODFET



## Carrier Density Profile (CV)

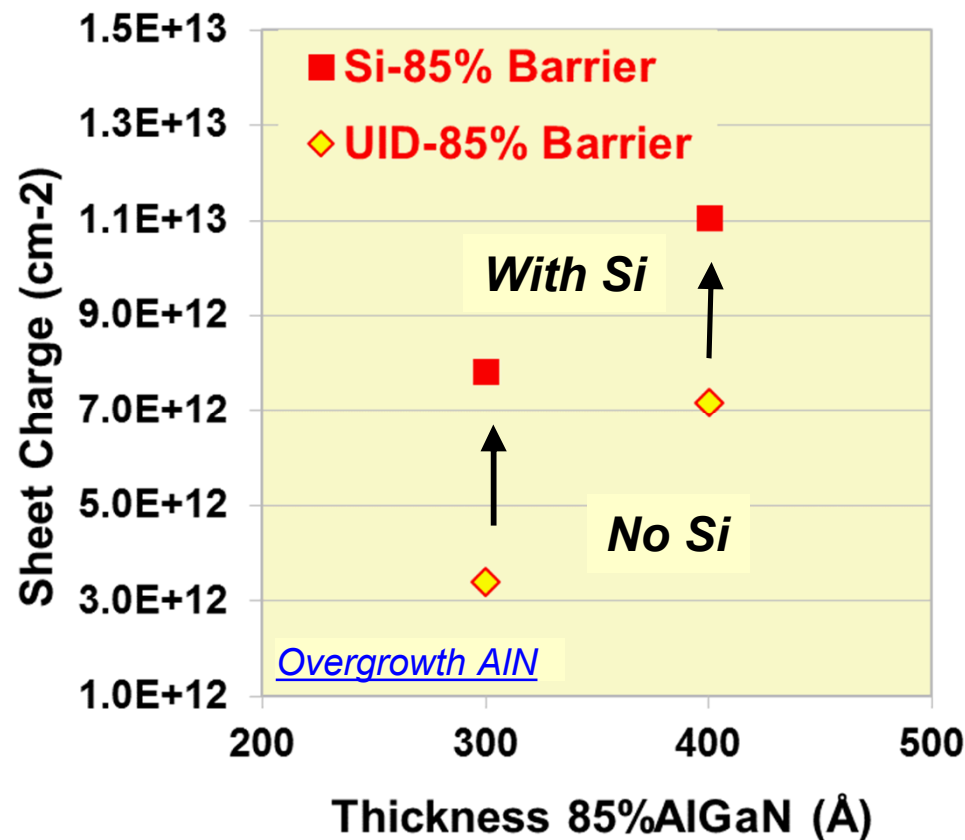


- No indication of parasitic channel
- Mobility is notably higher

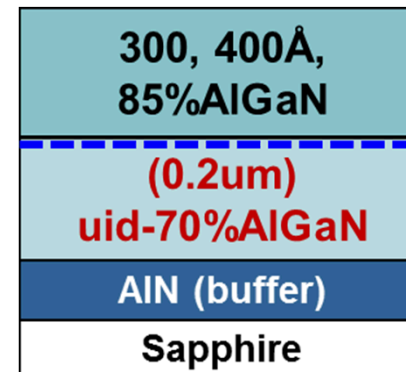
➔ 2DEG can be formed over the full range of compositions

# Si-doped and UID-barrier of 85% / 70% AlGa<sub>N</sub> heterostructure

*V(pinch-off) with & without  
Si doping of 85% Barrier*



*85% / 70% HEMT structure*



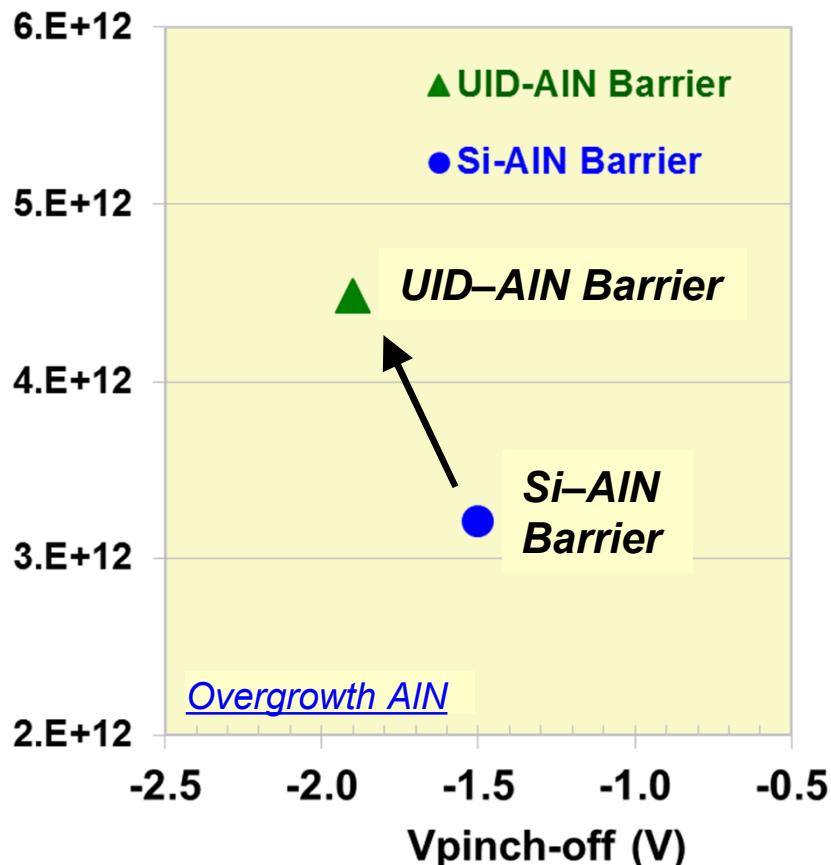
← Si-doped &  
UID-barrier

- ➡ 2DEG formation by polarization fields (No Si)
- ➡ Si adds charge to the channel



# Si-doped and UID-barrier of AlN / 85% AlGaN heterostructure

## Sheet Charge vs. $V_{\text{pinch-off}}$ with & without Doping of AlN Barrier



### Si-doped AlN

100Å Si-AlN
100Å UID-AlN
(0.5μm) uid-85%AlGaN
AlN (buffer)
Sapphire

### UID-AlN

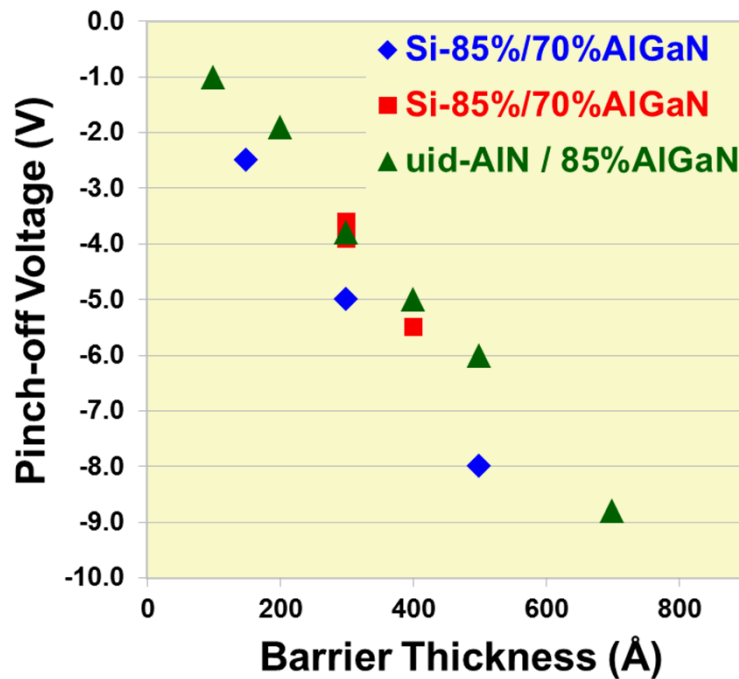
200Å UID-AlN
(0.5μm) uid-85%AlGaN
AlN (buffer)
Sapphire

### Si-doped vs. UID- AlN barrier

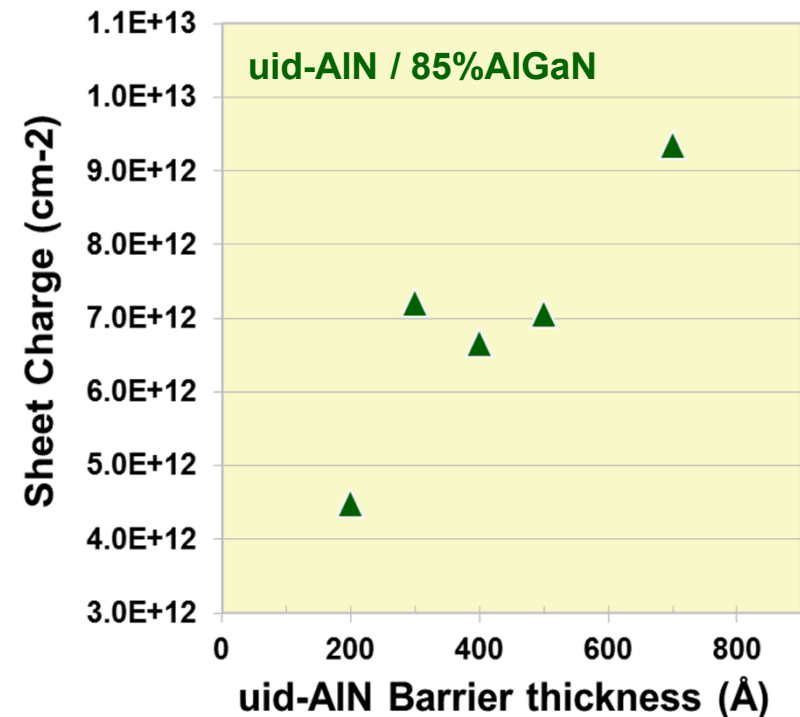
- 2DEG formation is very sensitive to Si-doping of AlN barrier.
- Sheet charge is higher w/o Si-doping
- 2DEG formed by polarization fields (NO Si doping).
- Formation of compensating defects in Si-AlN reduce sheet charge.

# Pinch-off voltage and sheet charge with barrier thickness

*V(pinch-off) vs.  
UID-AIN Barrier Thickness*



*Sheet Charge vs.  
UID-AIN Barrier Thickness*

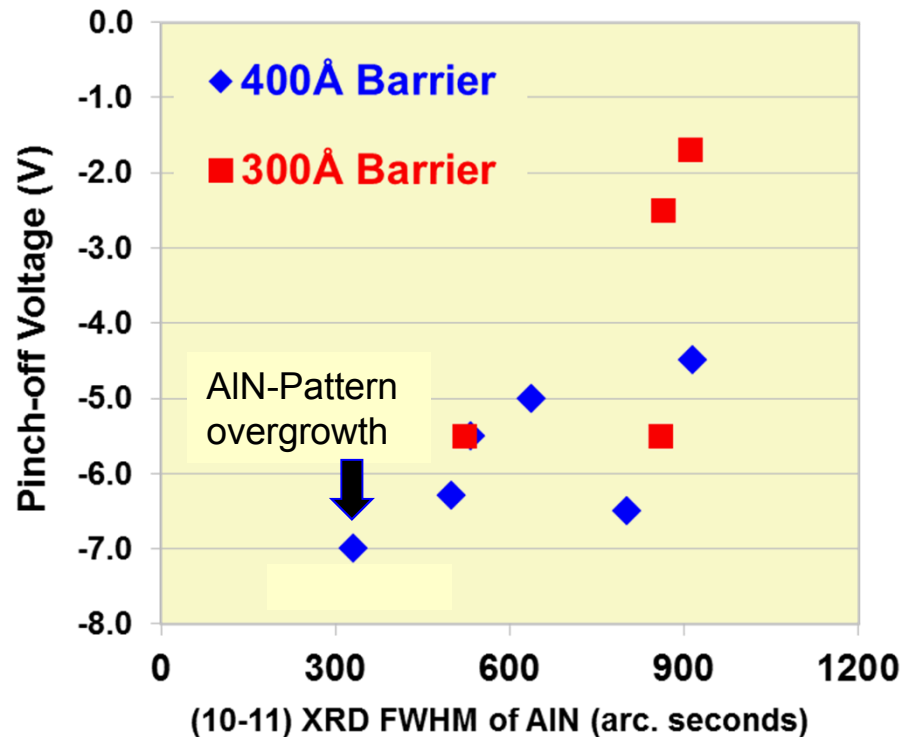


- Pinch-off voltage can be controlled with barrier thickness.
- No 2DEG with 1000Å barrier

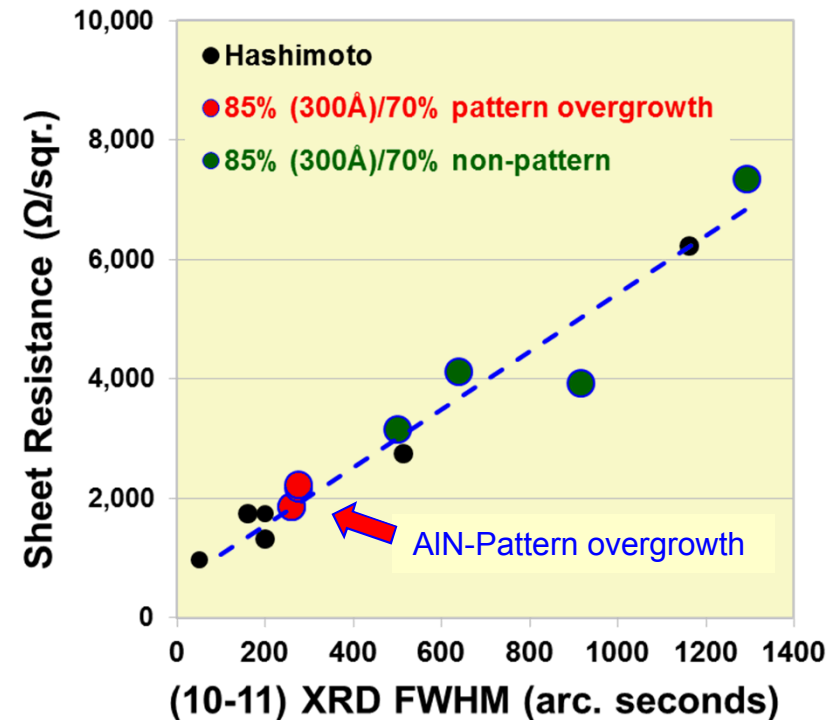
- $N_s$  approaching  $10^{13}$  (cm<sup>-2</sup>)
- Mobility: 157 – 289 (cm<sup>2</sup>/Vs)

# Pinch-off voltage and sheet charge of 85% / 70% AlGaN heterostructure vs. TDD

## $V(\text{pinch-off})$ vs. TDD of AlN



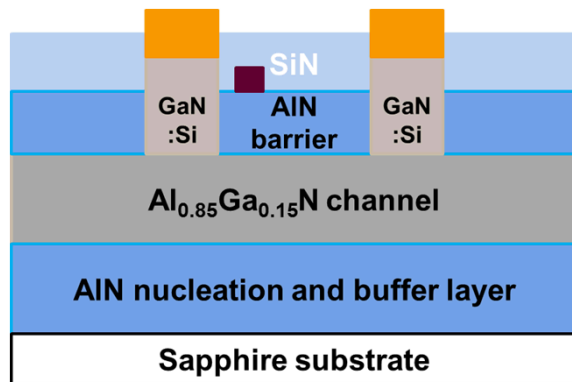
## Sheet Resistance ( $R_s$ ) vs. TDD of AlN



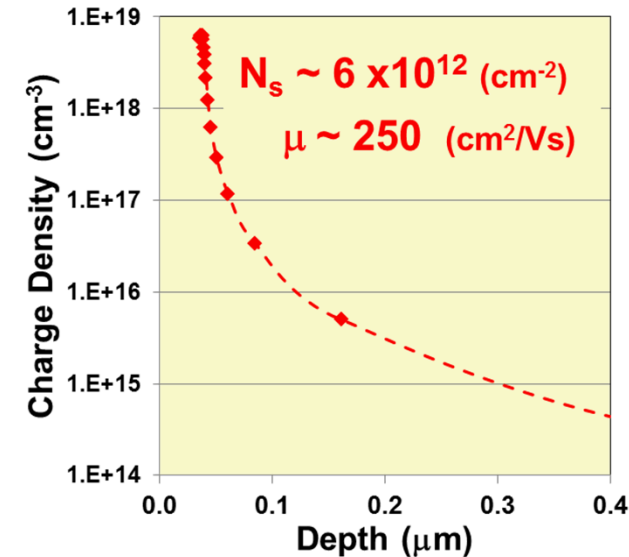
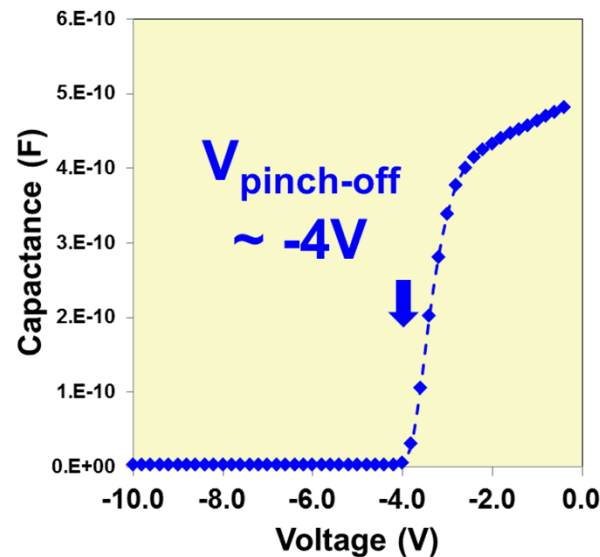
➡ Pinch-off voltage and sheet resistance depend on dislocation density

# HEMT device from an AlN / 85%AlGaN heterostructure

## AlN / 85%HEMT structure



## CV Characterization

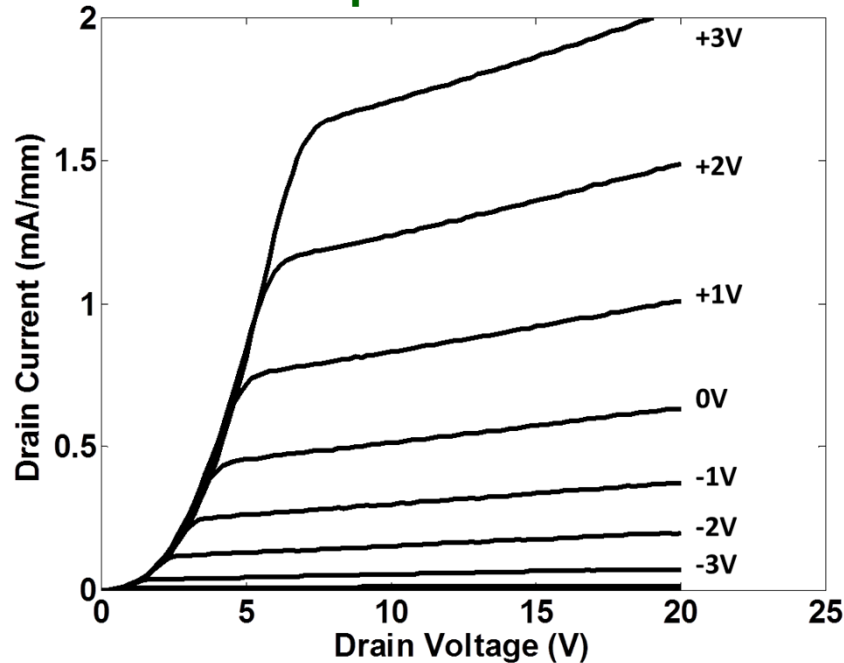


- AlN (475 Å) / 85%AlGaN (0.4 μm) heterostructure
- Gate-Drain spacing: 10 μm
- Regrown contacts (Si-GaN)

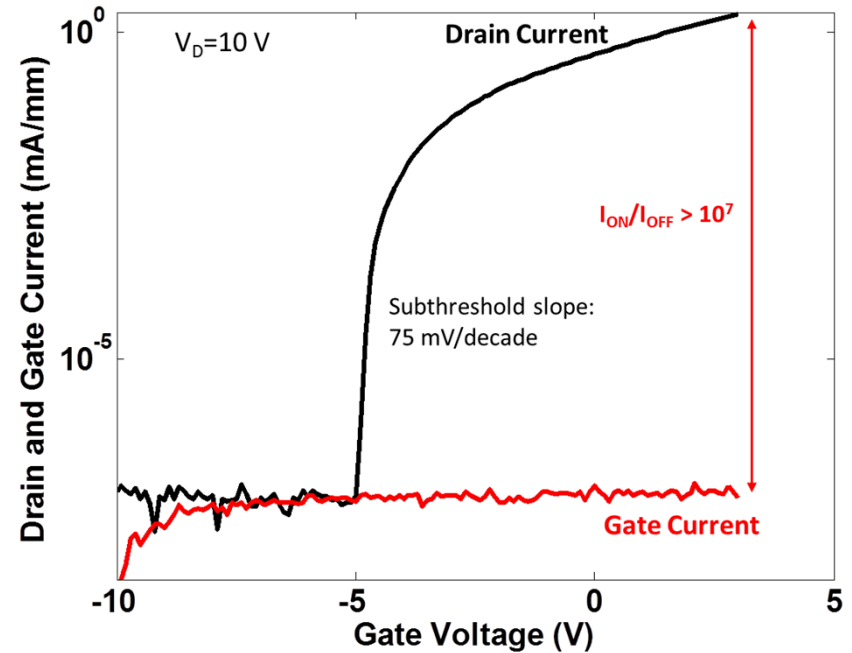
- Sheet resistance: 4200 ohm/sqr.
- Pinch-off voltage: -4 V
- Sheet charge density:  $6 \times 10^{12} \text{ cm}^{-2}$
- Inferred mobility: 250  $\text{cm}^2/\text{Vs}$

# HEMT characteristics based on an $\text{AlN} - \text{Al}_{0.85}\text{Ga}_{0.15}\text{N}$ heterostructure

## FET Operation



## On/Off State Characteristics



- Good pinch-off (-4.5V)
- Knee voltage linear with gate voltage
- Low drain current
- Contacts are not ohmic

- Low drain and gate leakage current
- Sub-threshold slope: 75mV/decade
- $I_{ON}/I_{OFF}$  ratio  $> 10^7$
- $V_{\text{breakdown}} = 810\text{ V}$  (no field plate)



- **Quasi-vertical  $\text{Al}_{0.3}\text{Ga}_{0.7}\text{N}$  PIN diodes**
  - Breakdown voltage of  $> 3000\text{V}$ , ( $E_c \sim 5.9\text{ MV/cm}$ , drift layer:  $11\mu\text{m}$ )
  - Low reverse leakage current (few nA) with TDD  $\sim 1\text{-}2\text{e}9\text{ cm}^{-2}$
  - High ON resistance ( $>20\text{ m}\Omega\text{-cm}^2$ ) due to poor lateral current spreading
- **2DEG formation in high Al ( $X_{\text{Al}} > 70\%$ ) AlGaIn heterostructures**
  - With and without Si doping of barrier layer
  - Mobility  $150\text{-}300\text{ cm}^2/\text{Vs}$ ,  $N_s \sim 1\text{x}10^{13}\text{ cm}^{-2}$
  - Similar sheet resistances AlN and sapphire substrates
- **Demonstrated AlGaIn-based HEMT operation with largest bandgap**
  - Low drain & gate leakage,  $I_{\text{ON}}/I_{\text{OFF}}$  ratio  $> 10^7$ ,  $V_b = 810\text{ V}$  (no field plate)
  - Low current due to rectifying contacts