



Growth and Electrical Properties of AlGaN-based PN Diodes and High-Electron-Mobility Transistors

Andrew Allerman, M. H. Crawford, A. M. Armstrong, A. G. Baca, J. R. Dickerson, M. P. King, R. J. Kaplar, and S.R. Lee

Sandia National Laboratories, Albuquerque, NM

Acknowledgements:

M. Smith, K. Cross, V. Abate and L. Alessi

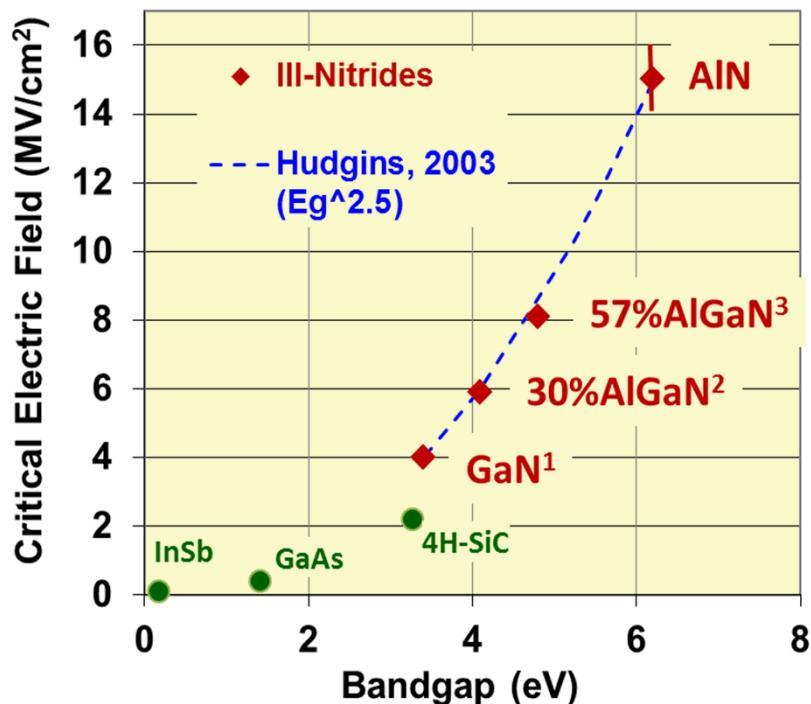
- Wide bandgap AlGaN alloys for power electronics
- PN diodes $\text{Al}_{0.3}\text{Ga}_{0.7}\text{N}$
 - Growth of thick, N- drift layers
 - Breakdown voltage and forward ON resistance
- Al-rich $\text{Al}_x\text{Ga}_{1-x}\text{N}$ ($x > 0.7$) heterostructures for HEMTs
 - 2DEG formation (growth conditions, doping & structure)
 - Initial transistor characteristics
- Summary

Ultra-wide-bandgap semiconductors (UWBS, >4eV) for power electronics

Critical Electric Field (MV/cm)

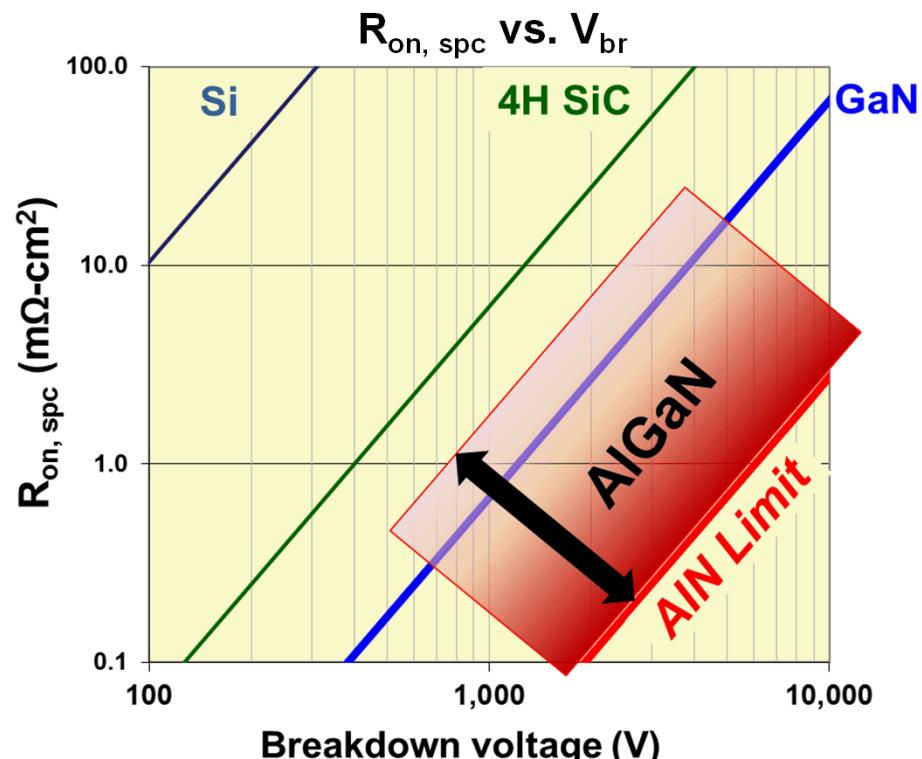
$$E_C \propto E_g^{2.5} \quad (\text{Direct gap})$$

Critical Electric Field vs. Bandgap



Unipolar Figure of Merit (vertical devices)

$$UFOM = \frac{V_{br}^2}{R_{on,sp}} = \frac{1}{4} \varepsilon \mu E_C^3 \propto E_g^{7.5}$$

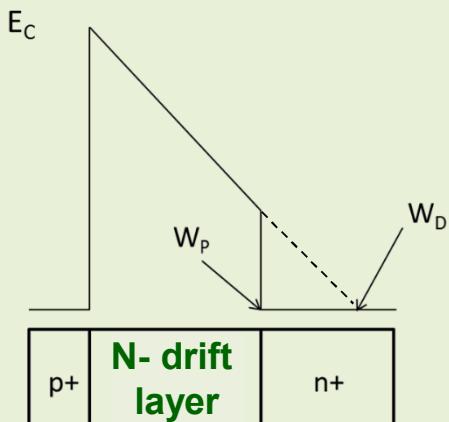


1-Armstrong, EL 2016, 2-Allerman, EL 2016, 3-Nisikawa, JpnJAP 2007

Breakdown voltage of drift layer: doping and thickness

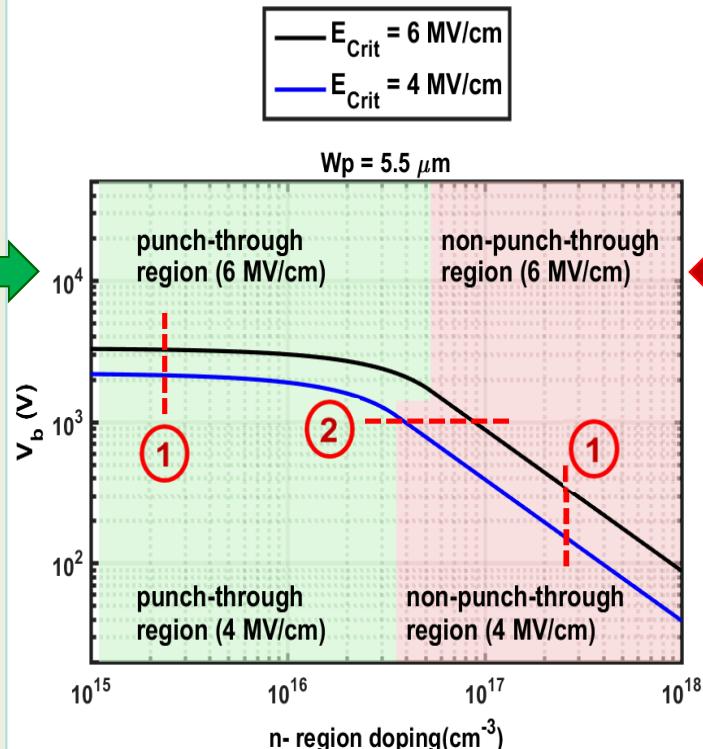
Punch-Through

$$W_P < W_D$$



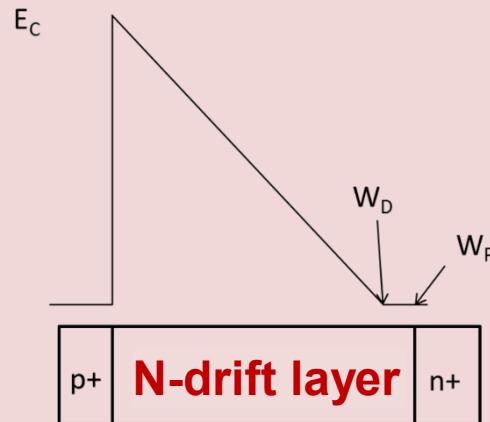
$$V_{br} = E_C W_P - \frac{q N_D W_P^2}{2\epsilon}$$

- N_d is lower than optimal
- ➔ Thickness determines V_{br}



Non-Punch-Through

$$W_P \geq W_D$$

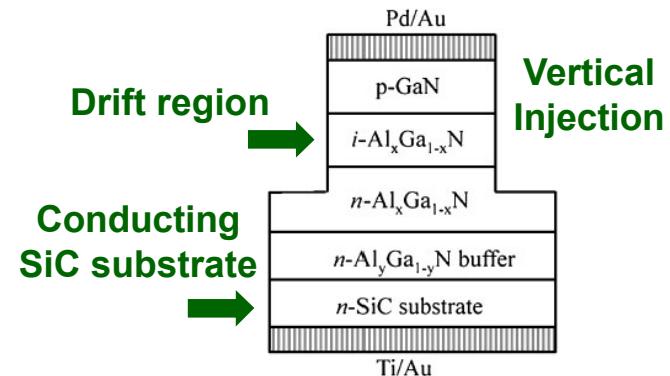


$$V_{br} = \frac{\epsilon E_C^2}{2qN_D}$$

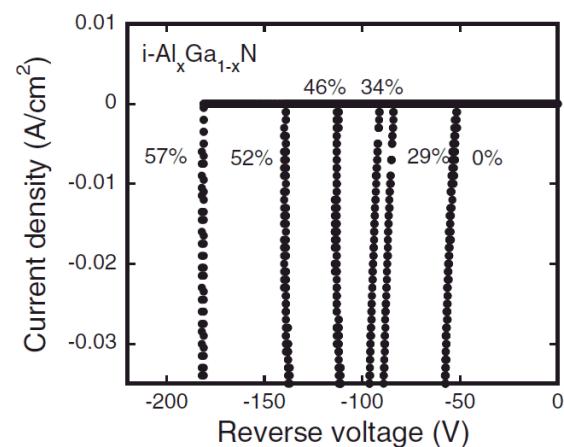
- Drift layer is thicker than optimal
- ➔ N_d determines V_{br}

Al_xGa_{1-x}N Vertical PN diode ($0 < X_{\text{Al}} < 0.57$)

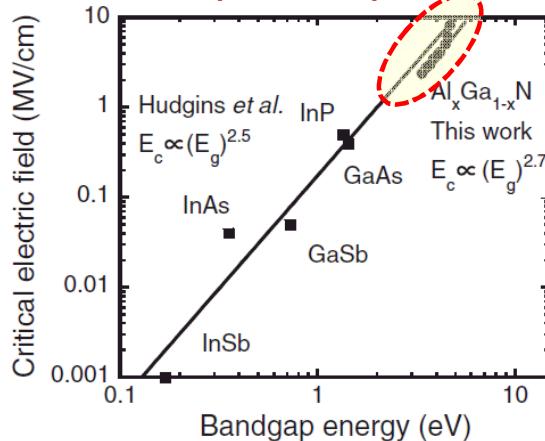
- Drift Layer: $\sim 0.2 \mu\text{m}$, $N_o \sim 2 \times 10^{16} \text{ cm}^{-3}$
- N-SiC substrates, $R_{\text{on,sp}} = 1.45 \text{ m}\Omega\text{-cm}^2$ ($X_{\text{Al}}=0.22$)



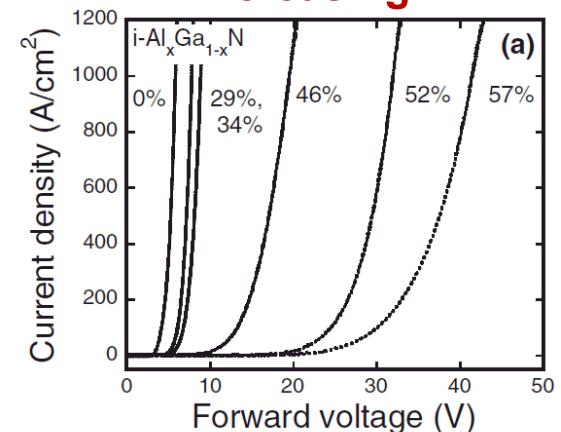
Reverse Breakdown
 $< 200\text{V}$



$E_{\text{crit}} \sim 8 \text{ MV/cm}$
(2x GaN)



Higher forward turn-on for increasing Al

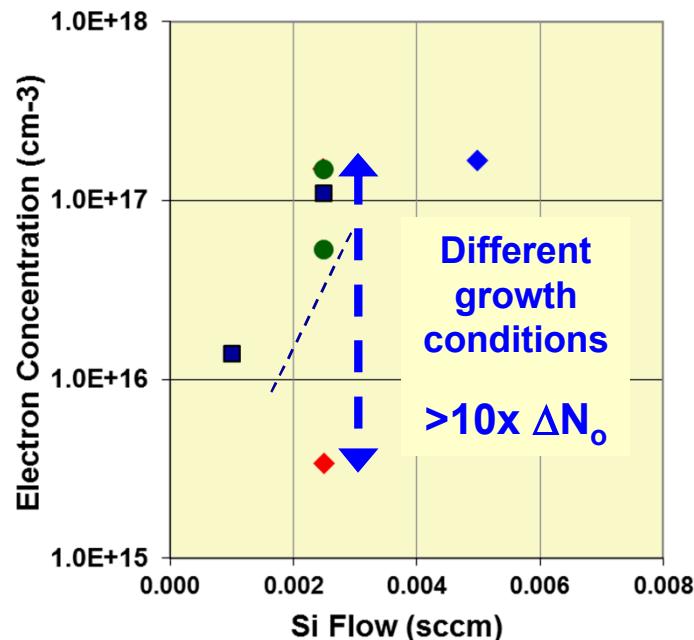


- ➔ Breakdown voltage increases with larger bandgap
- ➔ Critical electric field scales as $E_g^{2.7}$

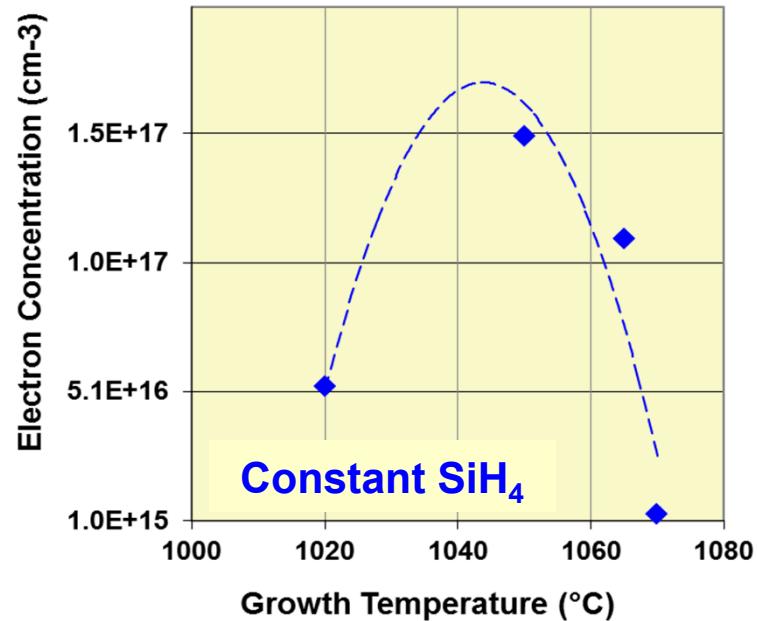
Control of N-type doping of drift region ($\text{Al}_{0.3}\text{Ga}_{0.7}\text{N}$)

→ *Si incorporation is linear in SiH_4 flow but...*

*Electron concentration (hall)
vs. SiH_4 flow (30%AlGaN)*

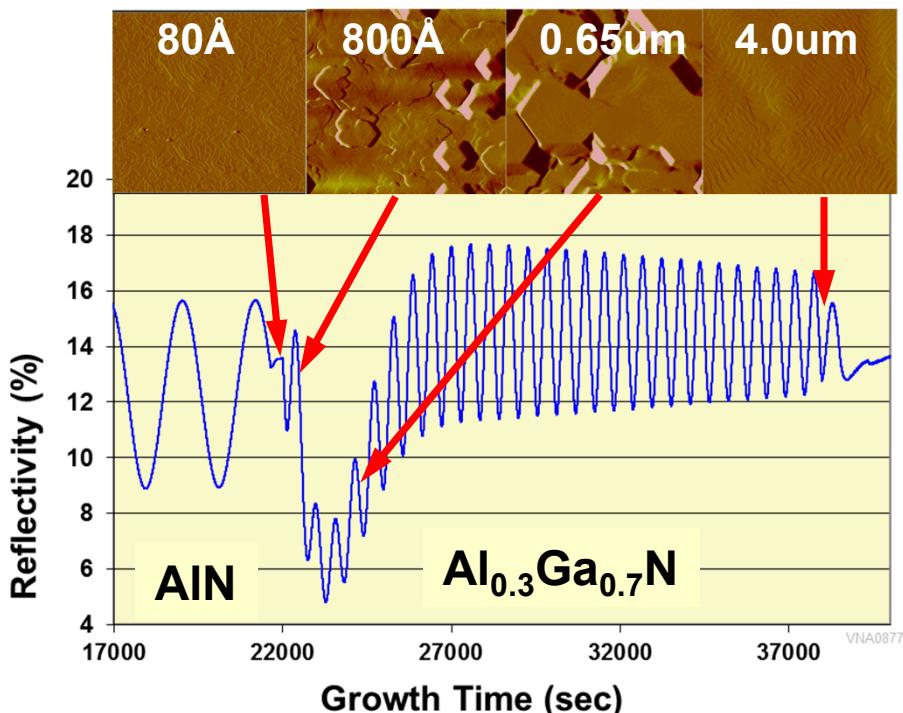


*Electron concentration (hall)
vs. growth temperature (30%AlGaN)*



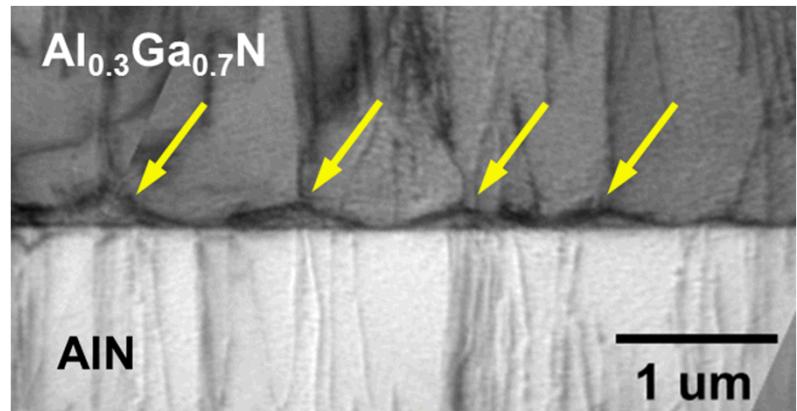
→ ... *electron concentration is highly dependent on
growth condition (i.e. the density of compensating defects)*

In-situ Reflectance and AFM



- Surface roughens (3D growth) and then planarization (2D growth)
- $\text{Al}_{0.3}\text{Ga}_{0.7}\text{N}$ layer is $>80\%$ relaxed, $\text{TDD} \sim 2-4 \times 10^9 \text{ cm}^{-3}$

Cross-section TEM



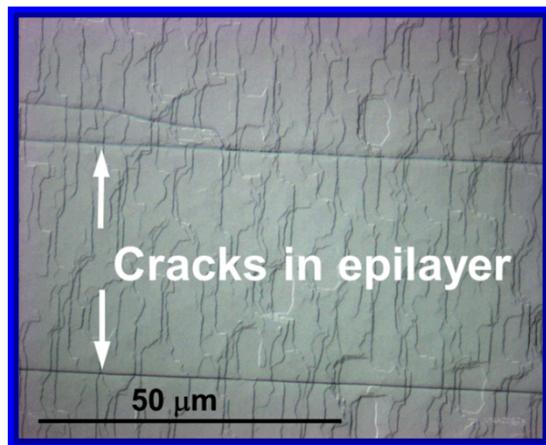
S. Lee, D. Follstaedt, J. Floro 2009

- Strain is relieved by interfacial misfit dislocation networks with hexagonal boundaries
- Threading dislocations “bundle” into triangular features

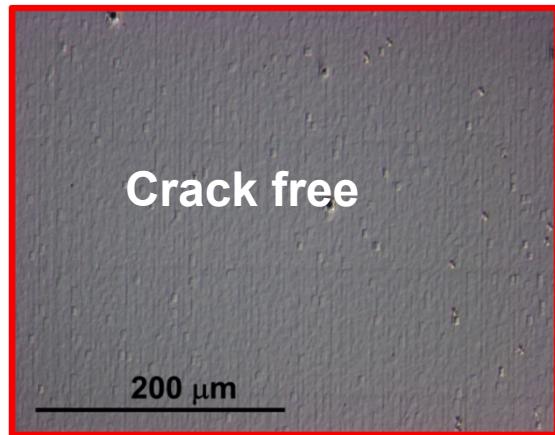
► **2D to 3D to 2D growth relieves strain & lowers TDD**

Reduce cracking and bow by using 3x thicker sapphire

Optical Image of $\text{Al}_{0.3}\text{Ga}_{0.7}\text{N}$ surface

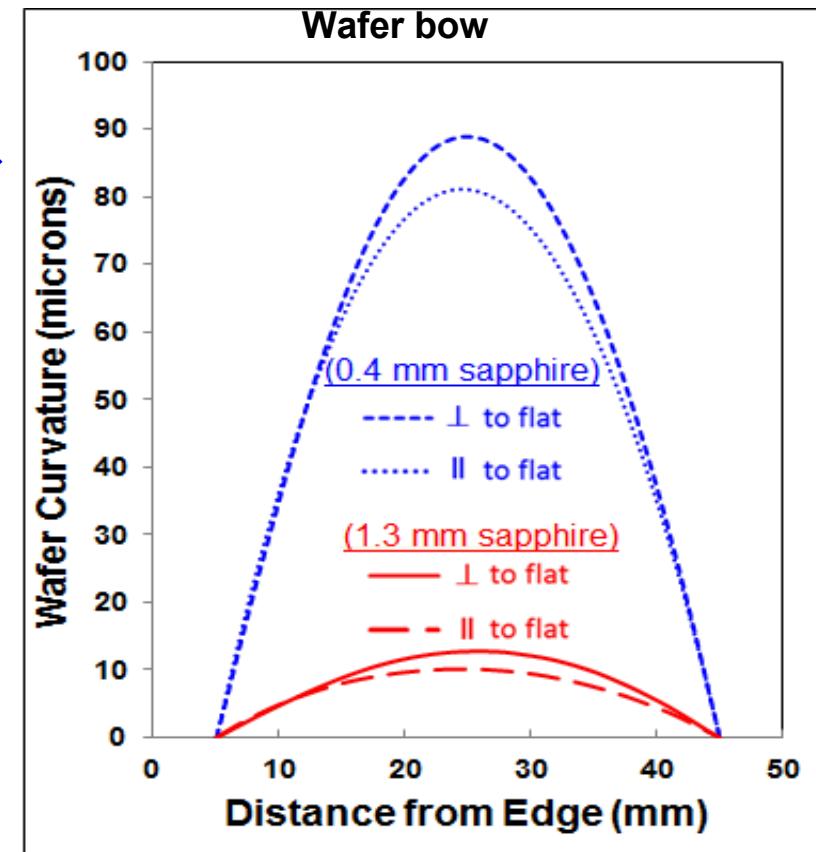


0.4 mm thick sapphire

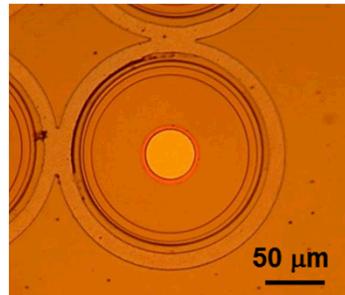


1.3 mm thick sapphire

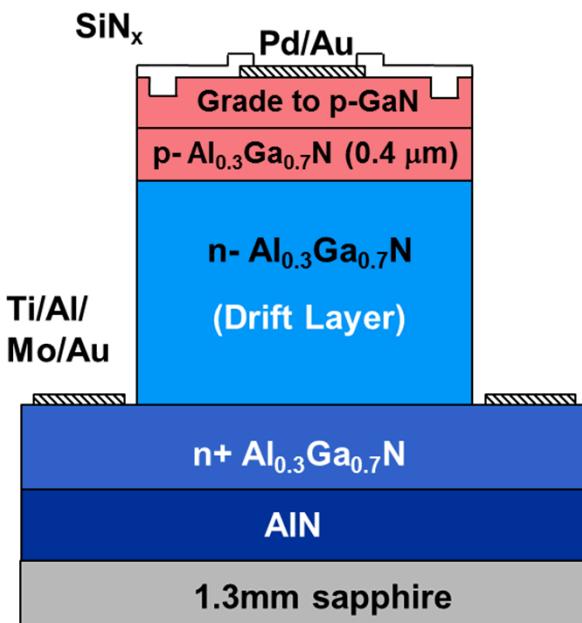
Cracking
& bowing



► 3x thicker sapphire reduces cracking & wafer bow



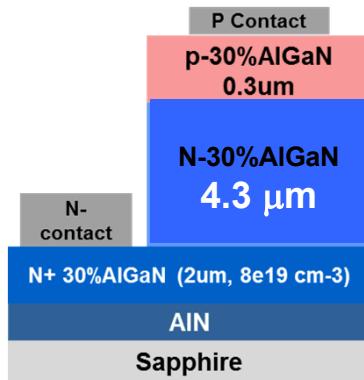
Optical image of diode



Quasi-Vertical PN diode structure

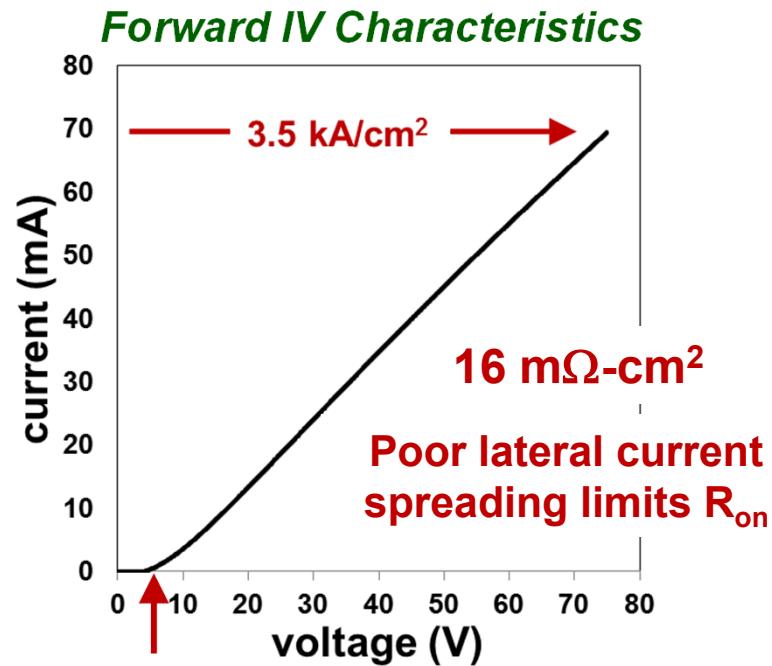
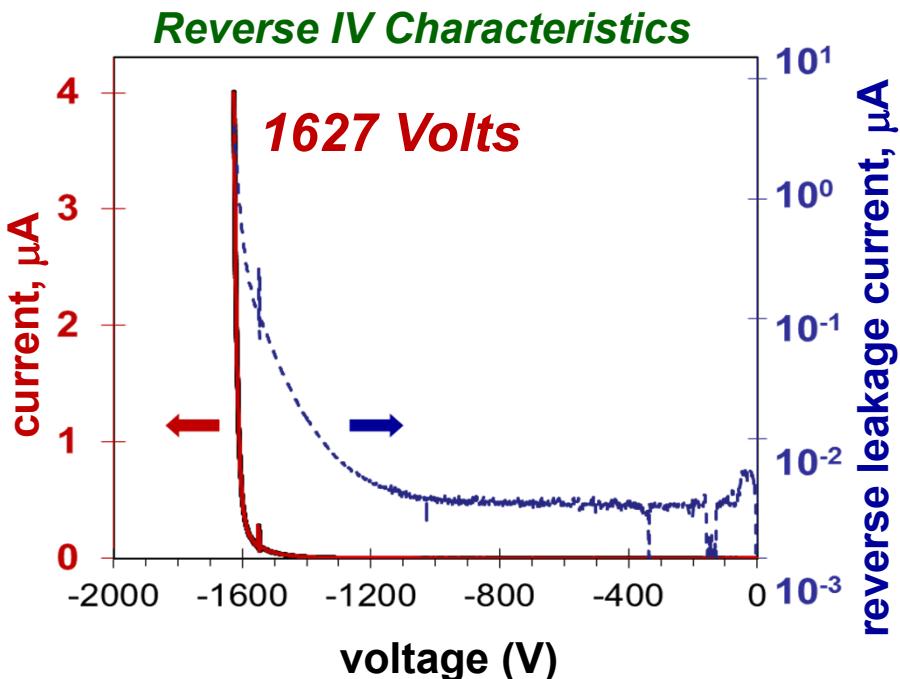
$\text{Al}_{0.3}\text{Ga}_{0.7}\text{N}$ PN diode design & fabrication

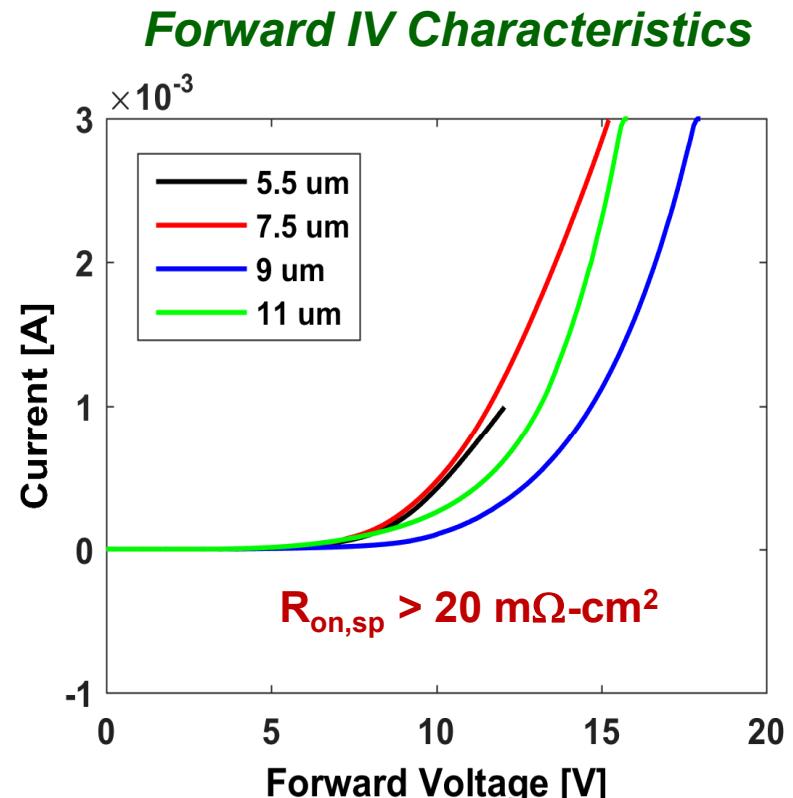
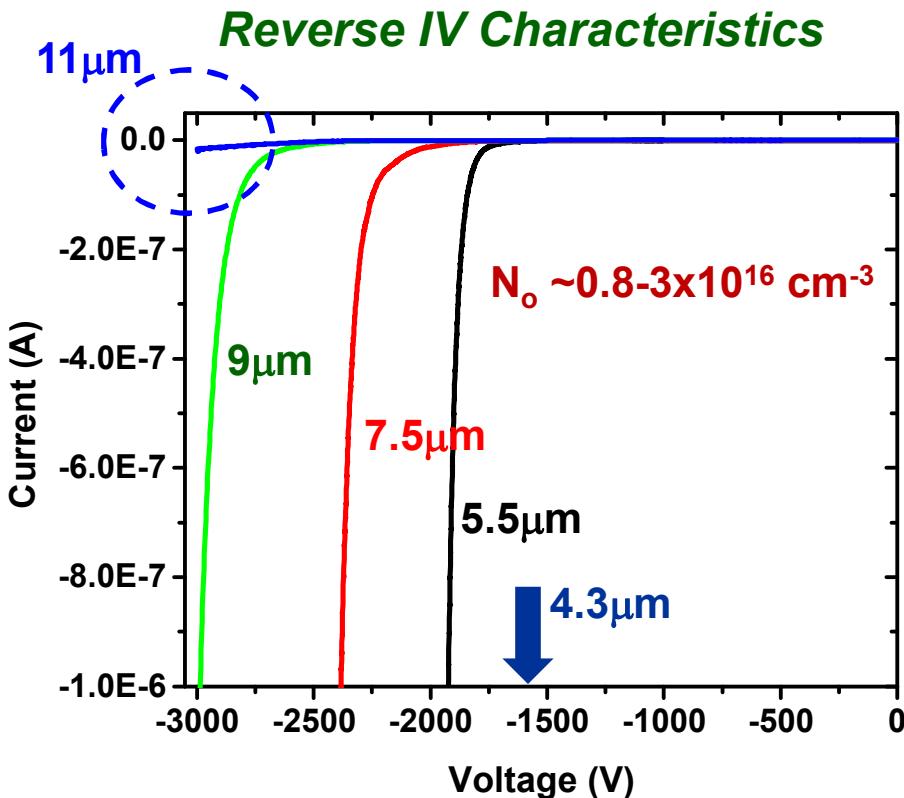
- Implanted junction edge termination
- Mesa / p-metal: 150 / 50 μm
- *Drift Layer: 4.3, 5.5, 7.5, 9, 11 and 15.5 μm*
- Crack-free (except 15.5 μm drift layer)
- Total epi thickness: 7 – 18, 22 μm
- Drift layer doping: mid 10^{15} – mid 10^{16} cm^{-3}
Mobility: 150 cm^2/Vs
- Sheet resistance: 40-80 Ω/sq .
- *Threading dislocation density: 1-3 x 10⁹ cm^{-2}*



- Drift layer: 4.3 μm , TDD $\sim 1-2\text{e}9 \text{ cm}^{-2}$
 $N_o \sim 5-7\text{e}16 \text{ cm}^{-2}$, $\mu \sim 150 \text{ cm}^2/\text{Vs}$,

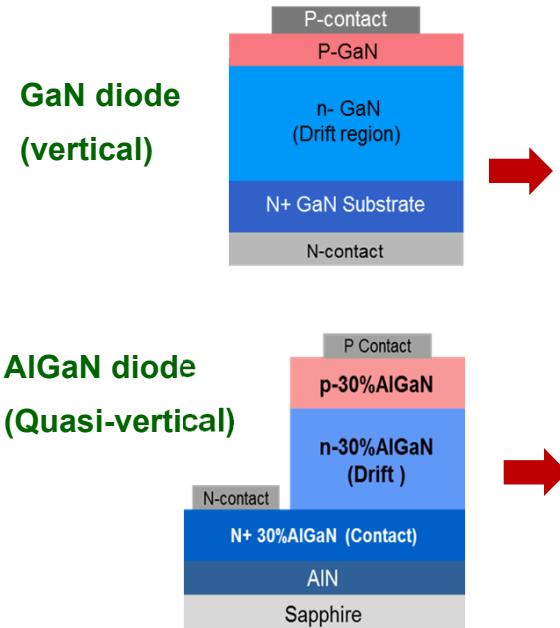
- ➡ $V_{br} > 1600 \text{ Volts}$
- ➡ $E_C \sim 5.9 \text{ MV/cm}$
- ➡ $I_{reverse} < 3\text{nA}$ (to 1200V)
- ➡ $V_{br}^2/R_{on,sp} = 150 \text{ MW/cm}^2$





- ➔ Breakdown voltages $\sim 3000\text{V}$ (Drift: 9 & 11 μm)
- ➔ V_{br} increases with drift layer thickness for N_o

Breakdown voltages reported for III-Nitride PN diodes



III-N PN diodes with > 3 kV breakdown voltage

Breakdown (kV)	No (cm-3)	Drift (μm)	Material	Group	Ref
4.7	2/9/16e15	33	GaN	Hosei Univ.	EDL 36p1180_2015
4.0	2-5e15	40	GaN	Avogy	EDL 36p1073_2015
3.9	3e15	30	GaN	Sandia	EL 52p1170_2016
3.7	5e15	>30	GaN	Avogy	EDL 35p247_2014
3.48	1/3/12e15	32	GaN	Hosei Univ.	IEDM15-237_2015
>3	0.8-3e16	11	30%-AGaN	Sandia	This work
3.0	0.8-3e16	9	30%-AGaN	Sandia	This work
3.0	1/10e15	20	GaN	Hitachi	Jpn J Appl Phys 52 p028007_2013

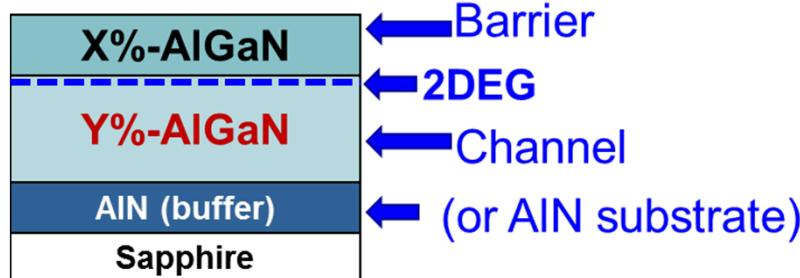
Advantages of wide-bandgap III-Nitride

	GaN	$\text{Al}_{0.3}\text{Ga}_{0.7}\text{N}$
N_o (cm ⁻³)	low e15	low e16
Drift (μm)	20-30	~10
TDD (cm ⁻²)	$\leq 1\text{e}6$	low 1e9

← Larger E_c (larger E_g)

← ??

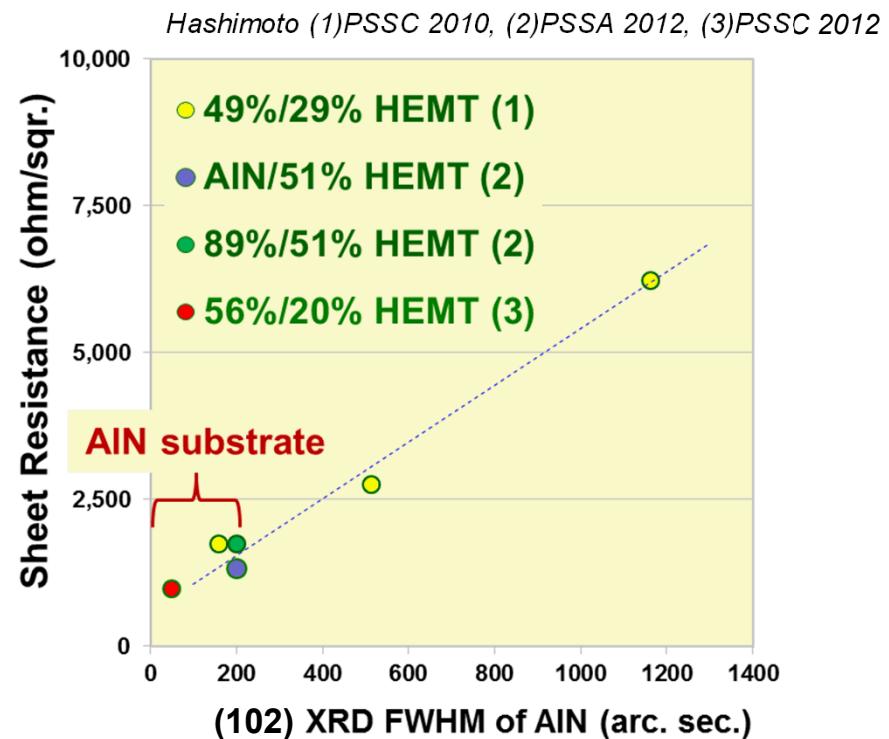
Typical heterostructure



Al-rich AlGaN Heterostructures

- N_s : $0.2 - 2.5 \times 10^{13} \text{ cm}^{-2}$
- Mobility: 140-170 (259, $X_{\text{al}}=0.2$) cm^2/Vs
- No 2DEG reported for channel $X_{\text{al}} > 0.6$
- Ohmic contacts are difficult
- Low threading dislocation density (TDD) is critical
- **AlN Substrates are expensive & small**

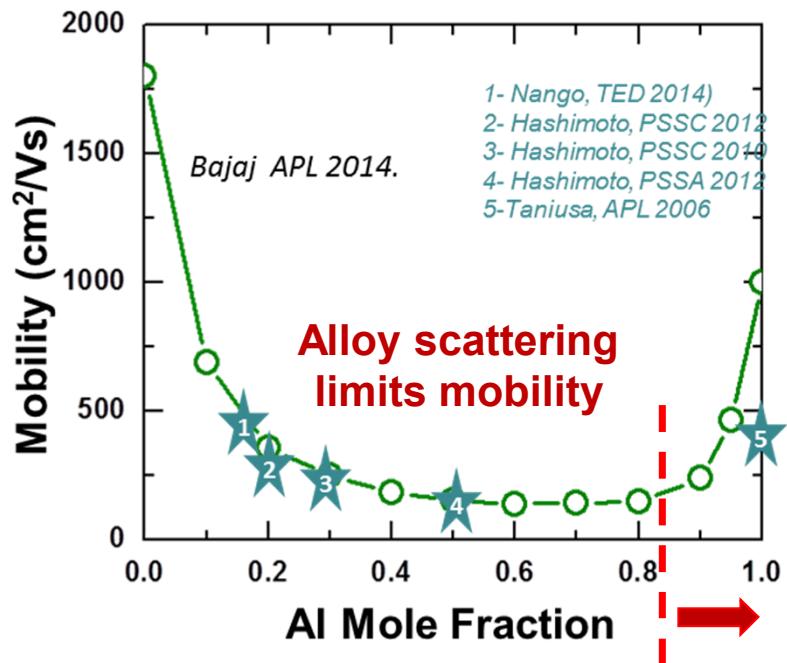
Sheet Resistance vs. Dislocation Density



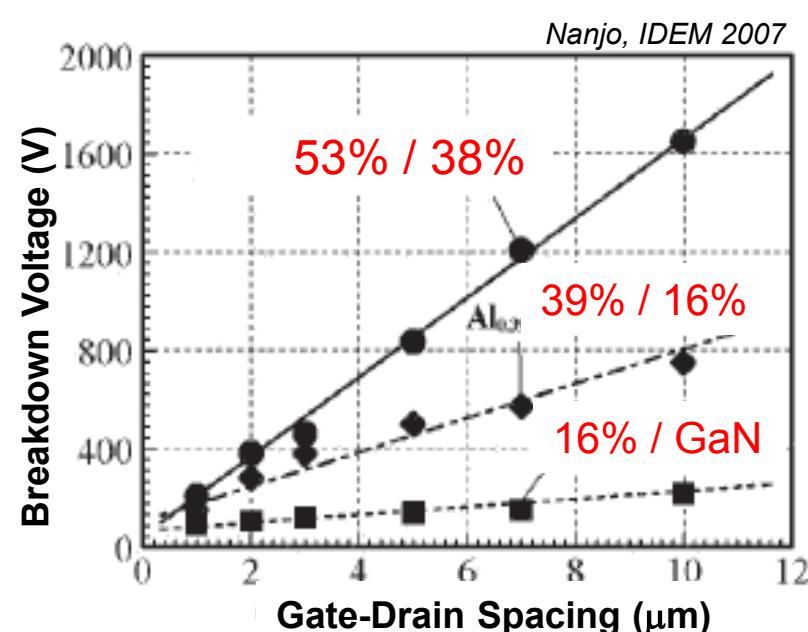
Q: Is there an alternative to AlN substrates?

Why higher Al compositions for AlGaN heterostructures?

Calculated Electron Mobility vs. AlGaN Channel Composition



Breakdown voltage of AlGaN HEMTs vs. G-D spacing



Higher Al compositions:

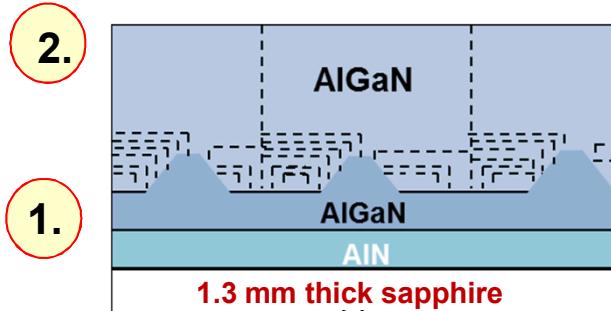
→ *higher mobility is predicted*

Higher Al compositions:

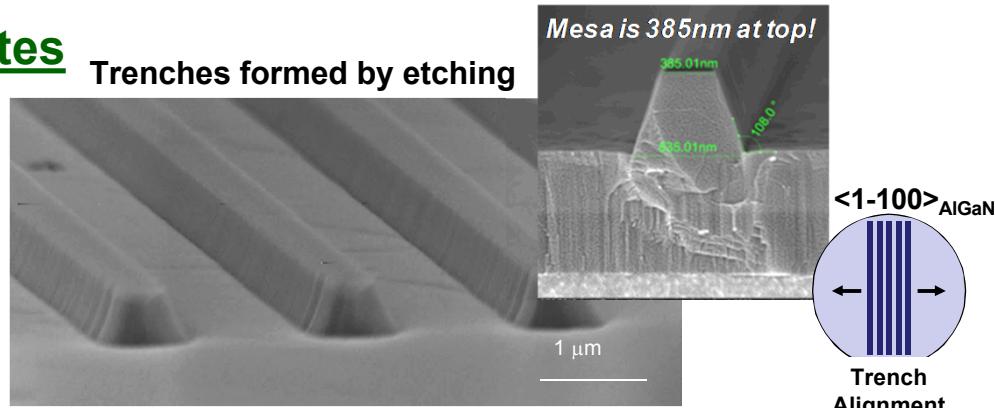
→ *higher breakdown voltages*

AlGaN Growth on Patterned Templates

AlGaN with reduced dislocations



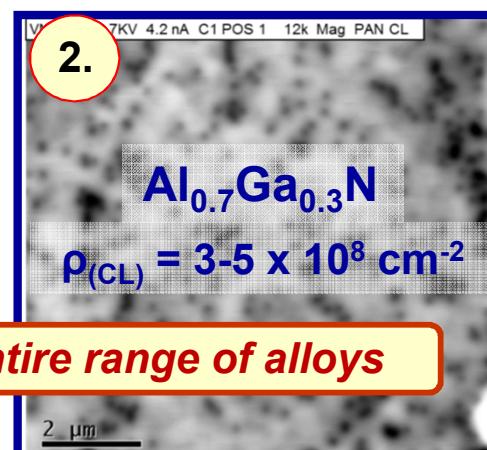
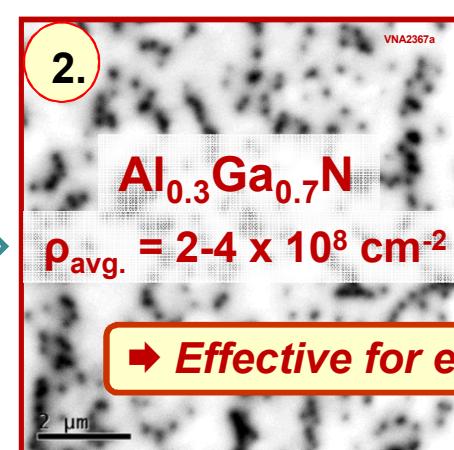
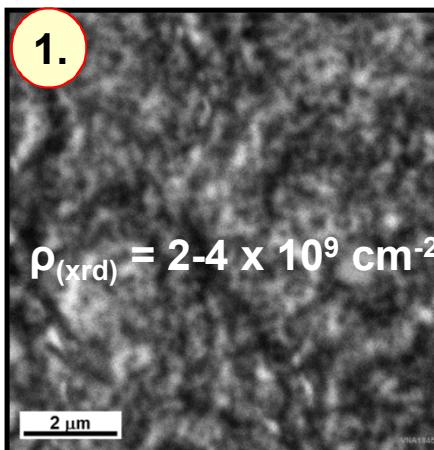
Trenches formed by etching



→ Sub-micron features are key for uniform reduction of dislocations

Allerman et. al., JCG 2014

Cathodoluminescence

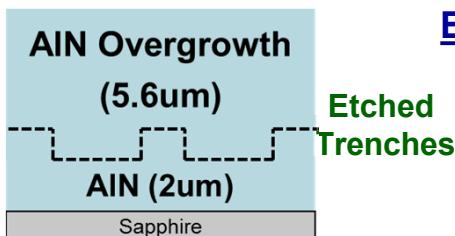


→ Effective for entire range of alloys

10-20X reduction

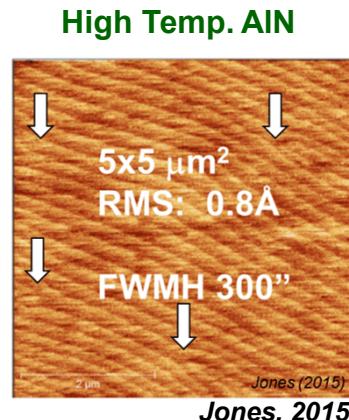
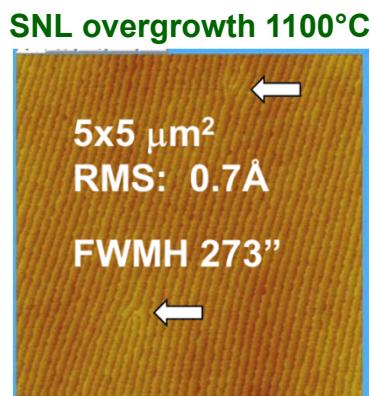
10-15x reduction

AlN overgrowth process (Allerman, JCG 2014)



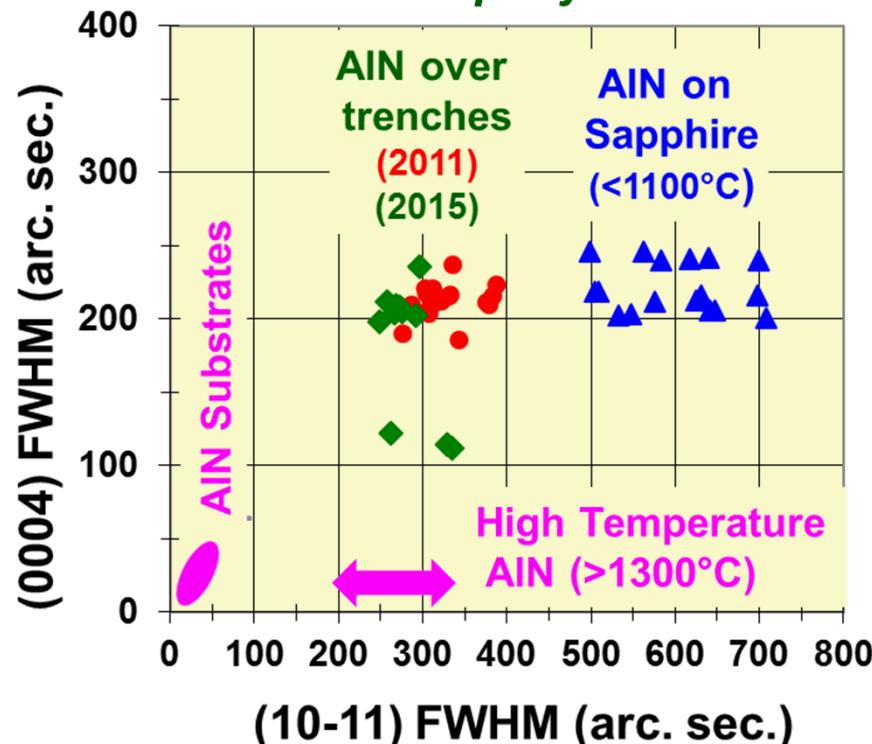
Etched Pattern

- 1x1 (mesa, trench, μm)
- 0.2 – 0.7 μm etch depth
- Overgrowth @ 1100°C



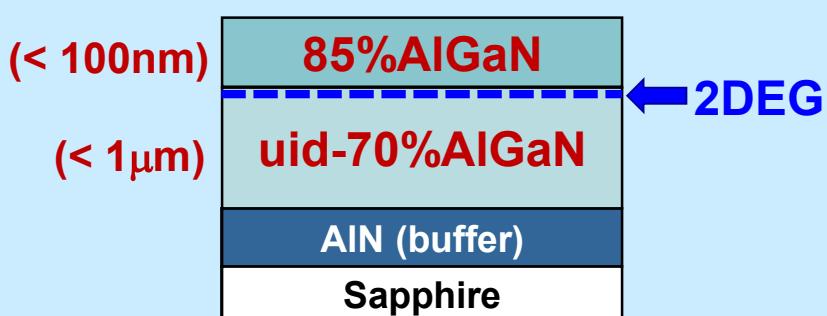
► SNL overgrowth @ 1100°C produces AlN similar to HT-AlN

X-ray diffraction peak – width of AlN epilayers



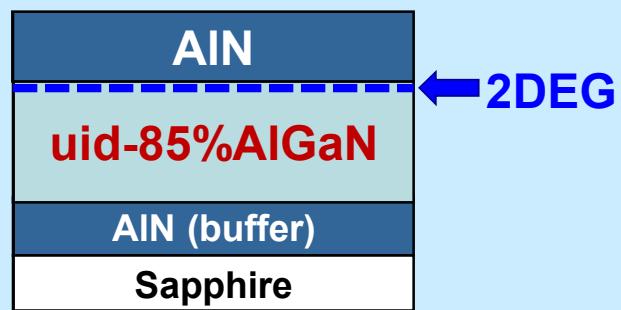
- AlN overgrowth: TDD $\sim 3\text{-}5 \times 10^8 \text{ cm}^{-2}$

85% / 70% AlGaN HEMT



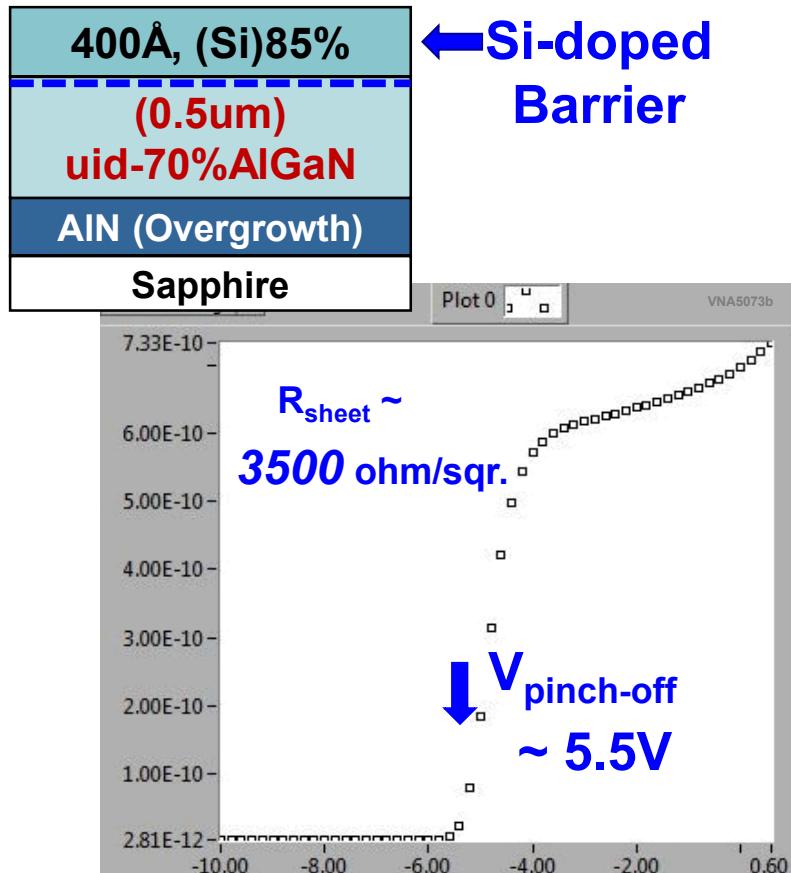
- $\text{Al}_{0.85}\text{Ga}_{0.15}\text{N}$ dopes N-type
- Expect lower mobility with lower Al compositions
- Structure could be grown pseudomorphic to AlN substrates

AlN / 85% AlGaN HEMT

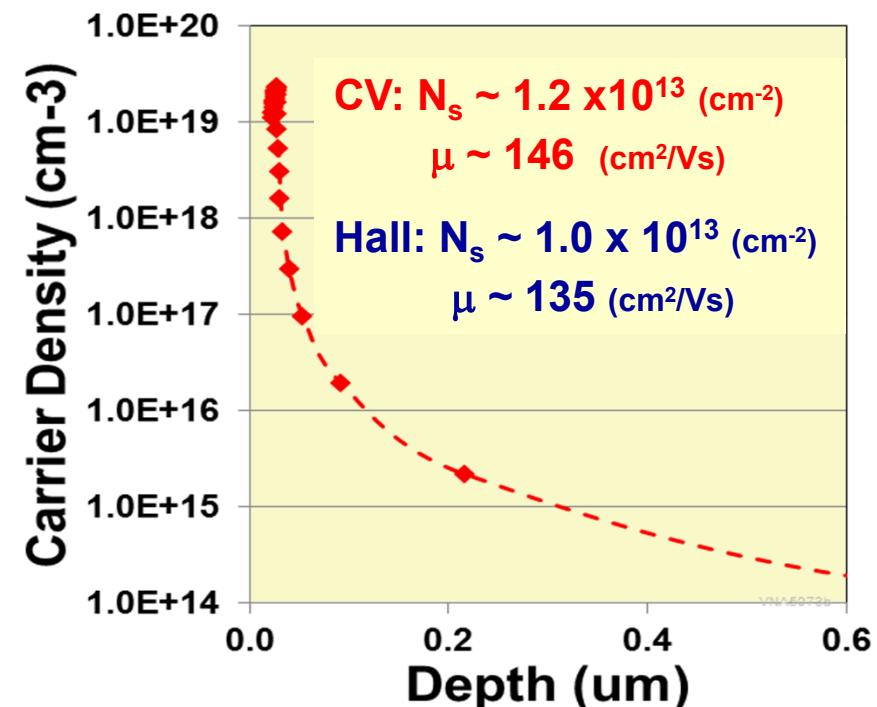


- AlN is hard to dope N-type
- Expect higher mobility with higher Al composition
- Well matched for pseudomorphic growth on AlN substrates

CV of 85% / 70% AlGaN MODFET



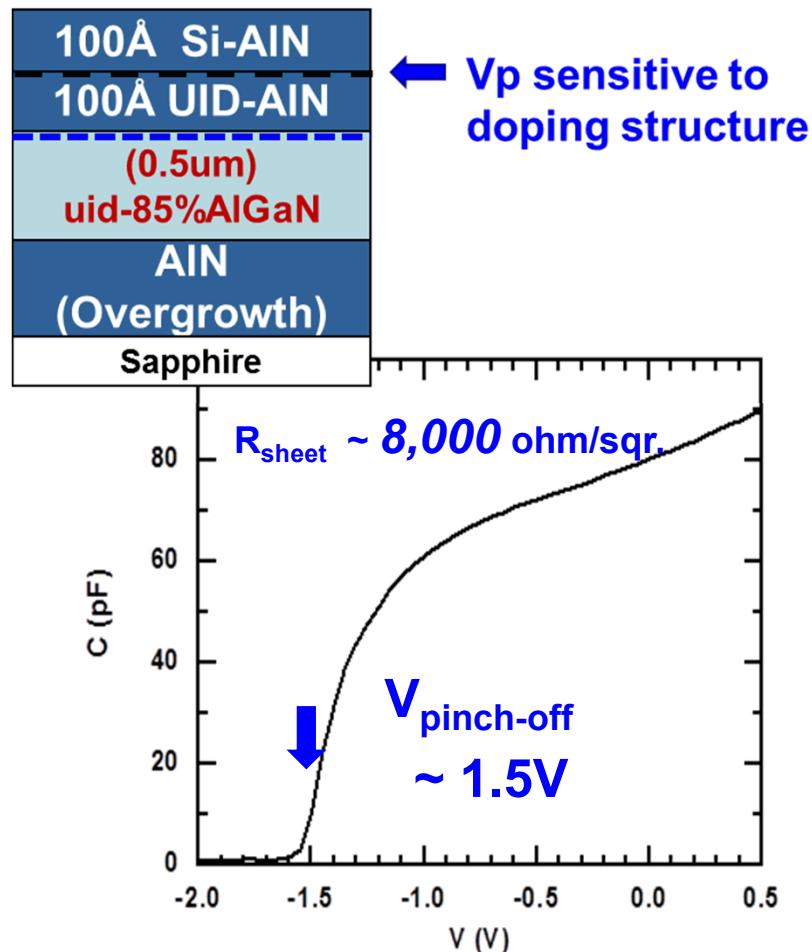
Carrier Density Profile (CV)



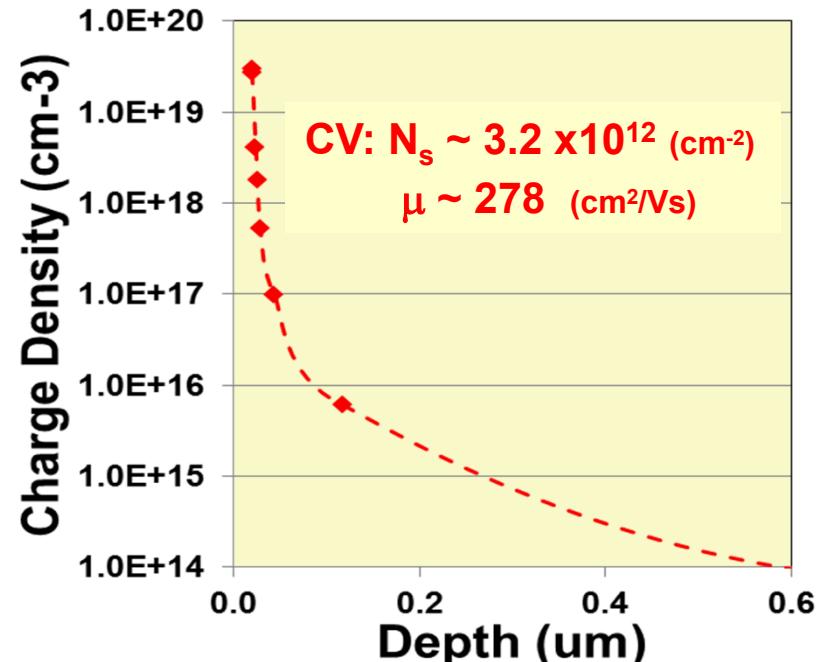
- N_s & μ are similar to HEMTs with lower Al channels

→ Demonstration of 2DEG in $\text{Al}_x\text{Ga}_{1-x}\text{N}$ channel for $x > 0.6$

CV of AlN / 85% AlGaN MODFET



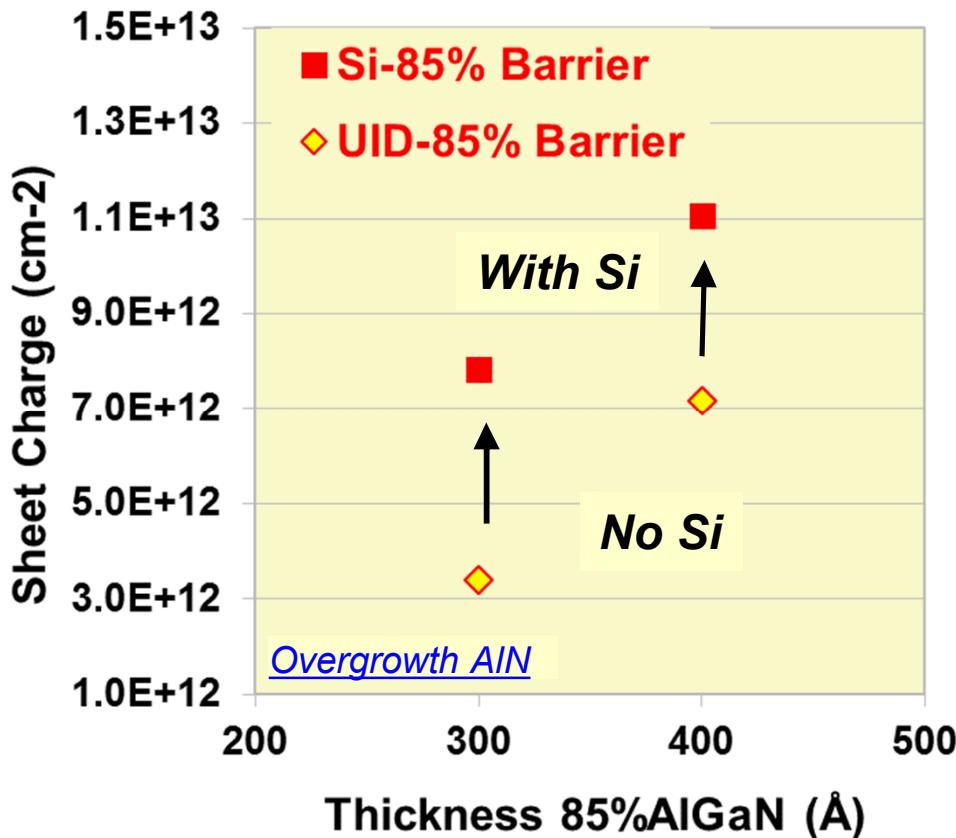
Carrier Density Profile (CV)



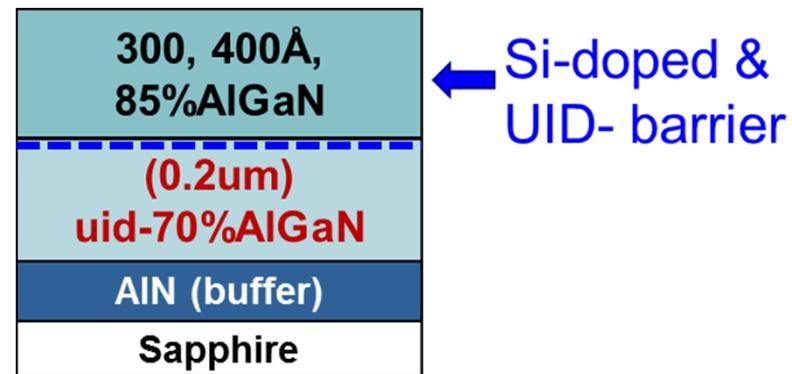
- No indication of parasitic channel
- Mobility is notably higher

► 2DEG can be formed over the full range of compositions

V(pinch-off) with & without Si doping of 85% Barrier

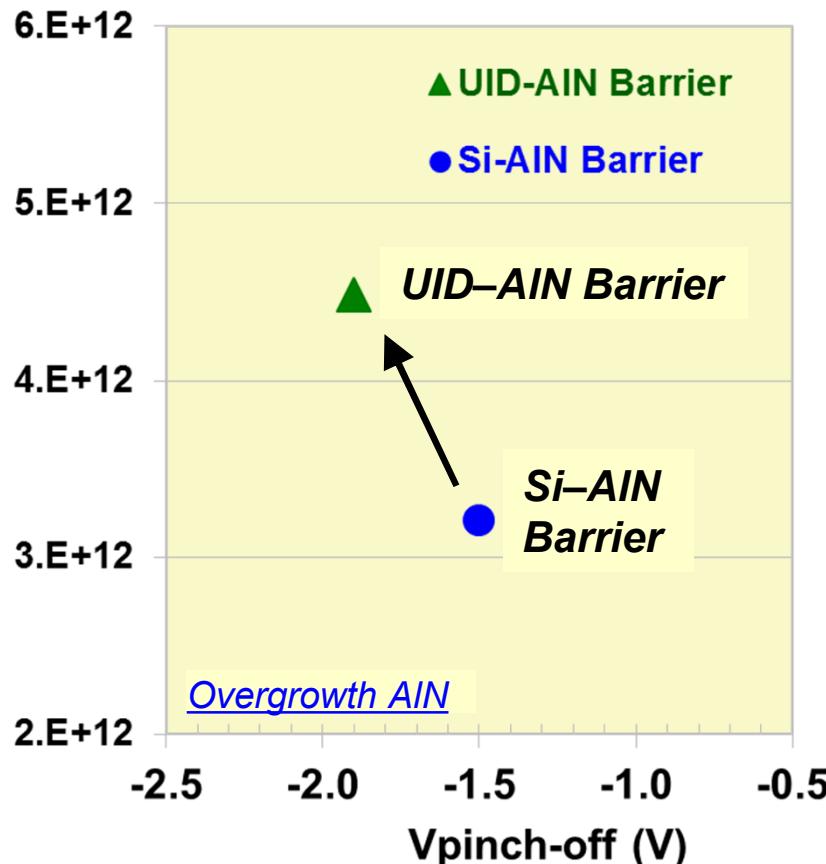


85% / 70% HEMT structure

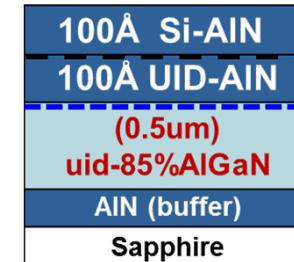


- 2DEG formation by polarization fields (No Si)
- Si adds charge to the channel

Sheet Charge vs. $V_{pinch-off}$ with & without Doping of AlN Barrier



Si-doped AlN



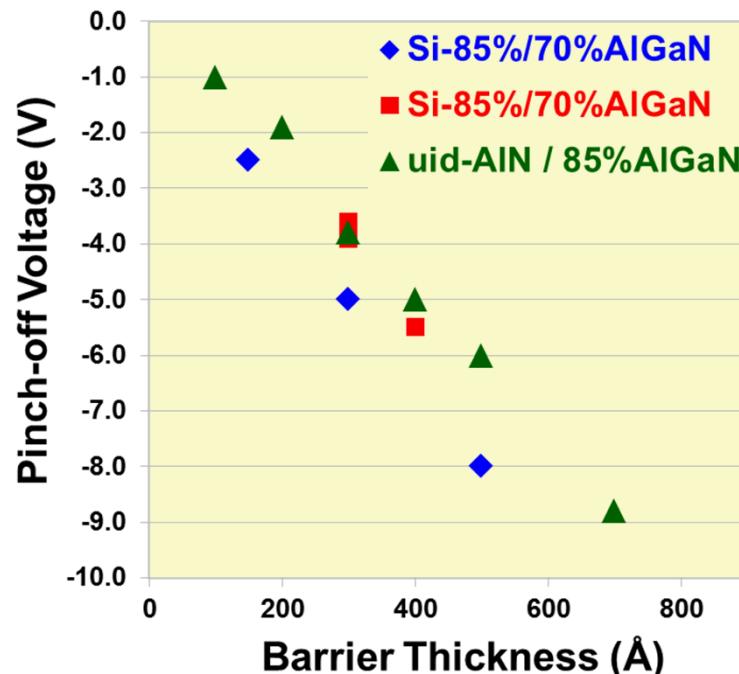
UID-AlN



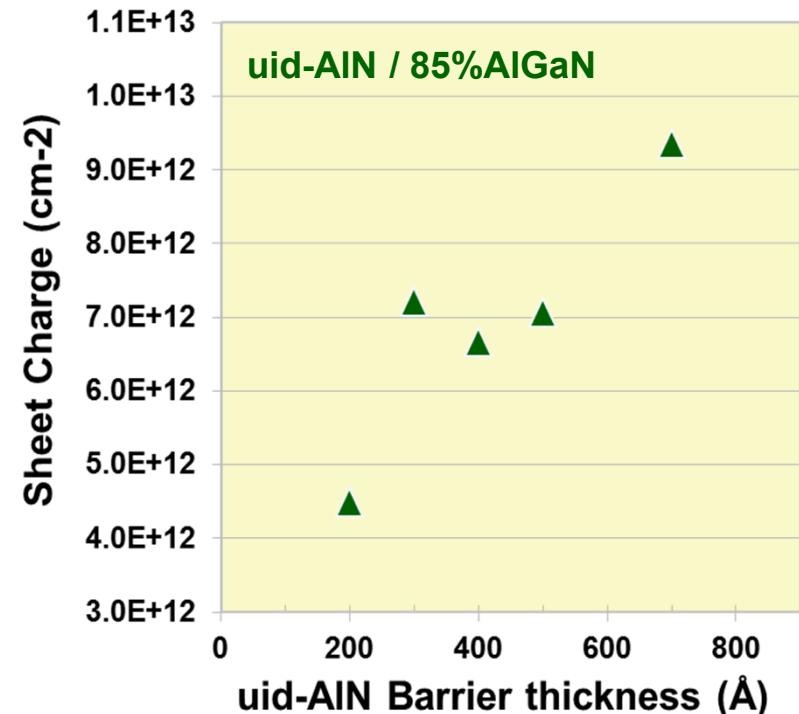
Si-doped vs. UID- AlN barrier

- 2DEG formation is very sensitive to Si-doping of AlN barrier.
- Sheet charge is higher w/o Si-doping
- 2DEG formed by polarization fields (NO Si doping).
- Formation of compensating defects in Si-AlN reduce sheet charge.

V(pinch-off) vs. UID-AlN Barrier Thickness



Sheet Charge vs. UID-AlN Barrier Thickness

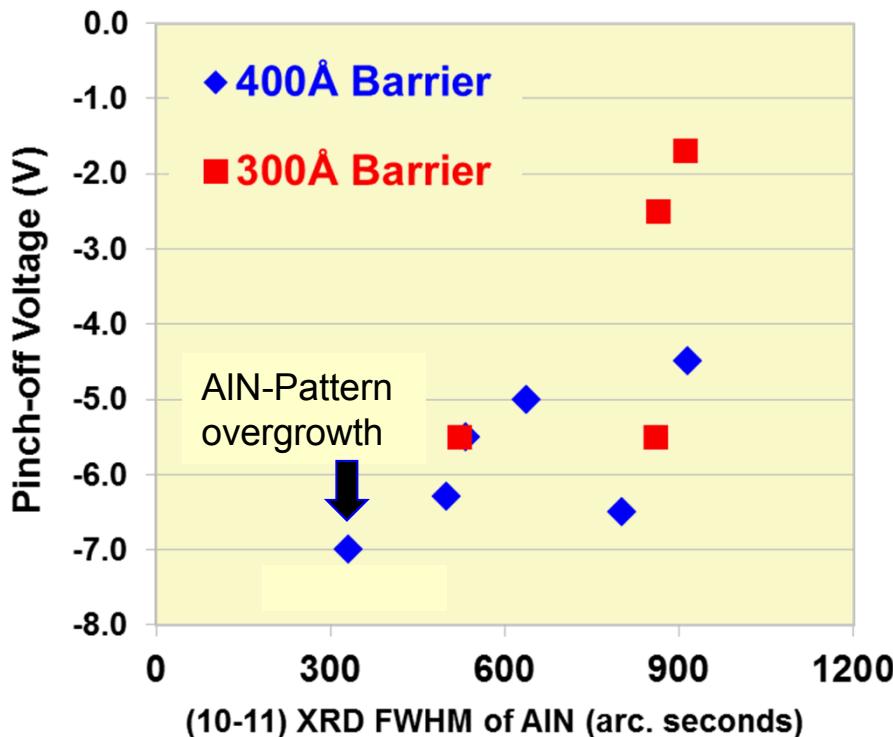


- Pinch-off voltage can be controlled with barrier thickness.
- No 2DEG with 1000Å barrier

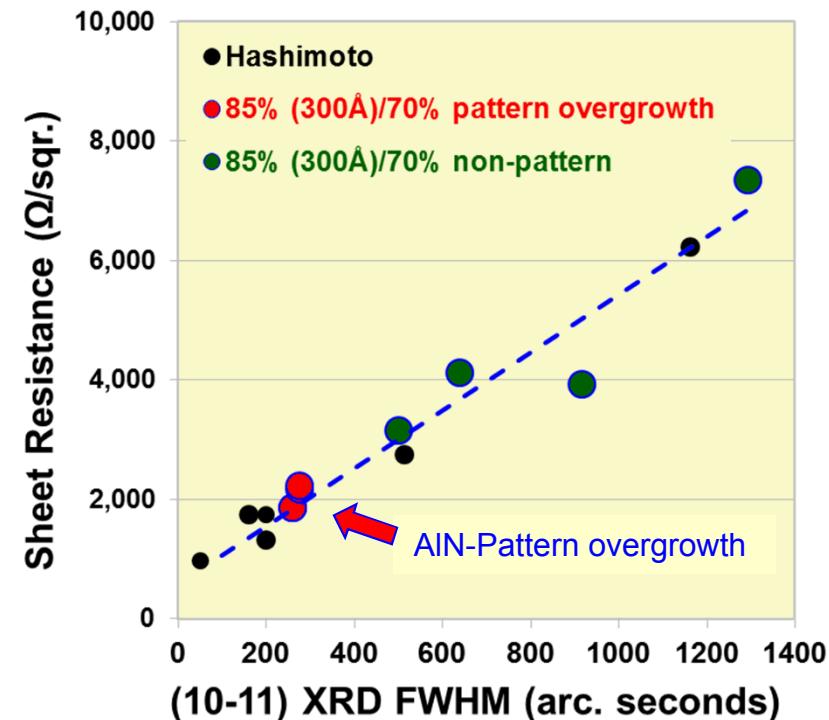
- N_s approaching 10^{13} cm^{-2}
- Mobility: $157 - 289 \text{ (cm}^2/\text{Vs)}$

Pinch-off voltage and sheet charge of 85% / 70% AlGaN heterostructure vs. TDD

V(pinch-off) vs. TDD of AlN

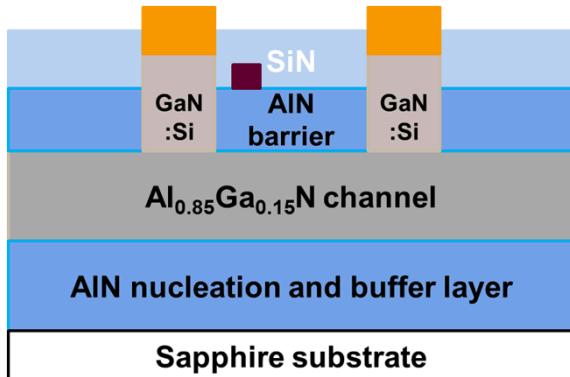


Sheet Resistance (R_s) vs. TDD of AlN

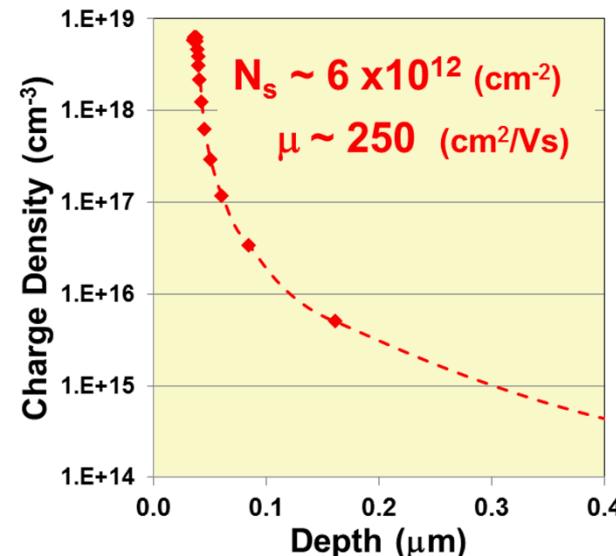
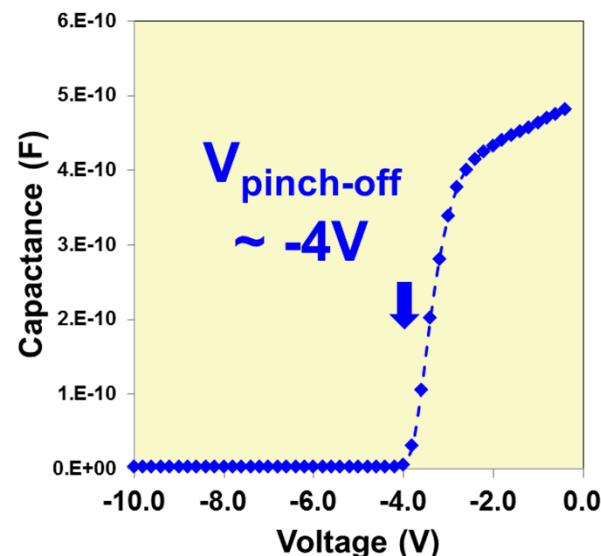


► Pinch-off voltage and sheet resistance depend on dislocation density

AlN / 85%HEMT structure



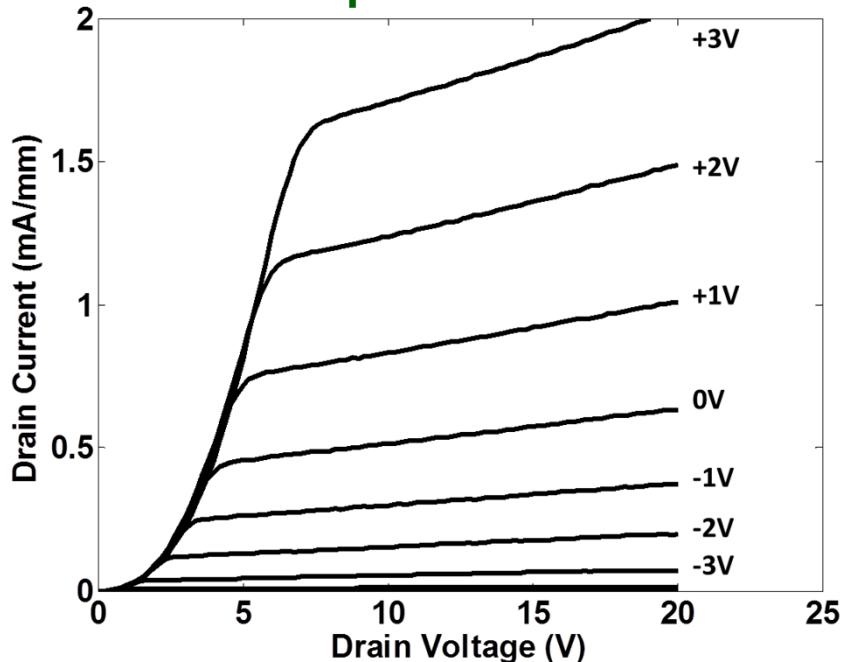
CV Characterization



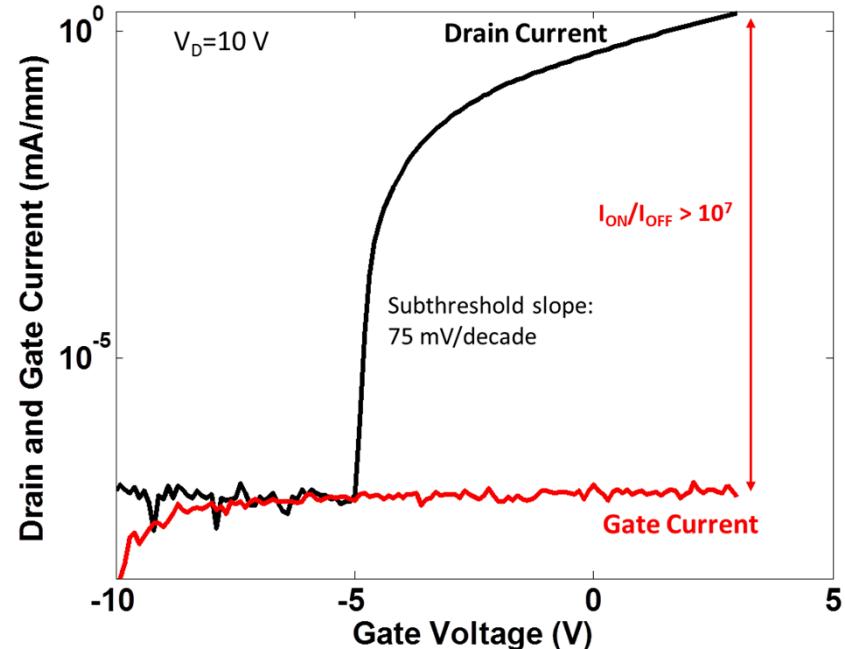
- uid-AlN (475 Å) / 85%AlGaN (0.4 μm) heterostructure
- Gate-Drain spacing: 10 μm
- Regrown contacts (Si-GaN)

- Sheet resistance: 4200 ohm/sqr.
- Pinch-off voltage: -4 V
- Sheet charge density: $6 \times 10^{12} \text{ cm}^{-2}$
- Inferred mobility: 250 cm²/Vs

FET Operation



On/Off State Characteristics



- Good pinch-off (-4.5V)
- Knee voltage linear with gate voltage
- Low drain current
- Contacts are not ohmic

- Low drain and gate leakage current
- Sub-threshold slope: 75mV/decade
- I_{ON}/I_{OFF} ratio $> 10^7$
- $V_{breakdown} = 810$ V (no field plate)

- Quasi-vertical Al_{0.3}Ga_{0.7} N PIN diodes
 - Breakdown voltage of > 3000V, (E_c ~ 5.9 MV/cm, drift layer: 11 μm)
 - Low reverse leakage current (few nA) with TDD ~ 1-2e9 cm⁻²
 - High ON resistance (>20 mΩ·cm²) due to poor lateral current spreading
- 2DEG formation in high Al (X_{al} > 70%) AlGaN heterostructures
 - With and without Si doping of barrier layer
 - Mobility 150-300 cm²/Vs, N_s ~ 1x10¹³ cm⁻²
 - Similar sheet resistances AlN and sapphire substrates
- Demonstrated AlGaN-based HEMT operation with largest bandgap
 - Low drain & gate leakage, I_{ON}/I_{OFF} ratio > 10⁷, V_b = 810 V (no field plate)
 - Low current due to rectifying contacts