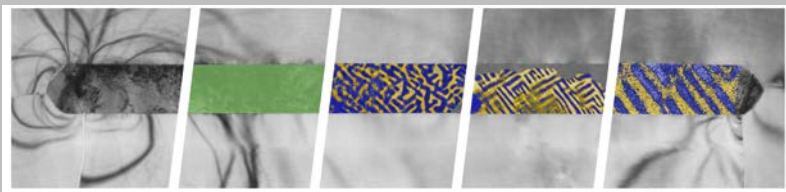
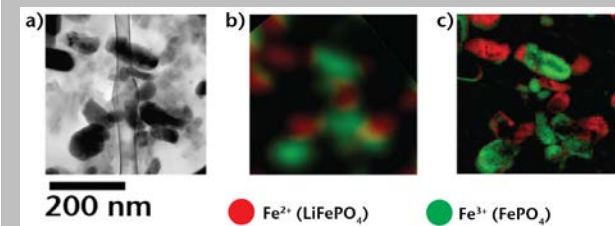


Exceptional service in the national interest



Aging Time →



Using Energy-Filtered TEM to Solve Practical Materials Problems with Inspirations from Gareth Thomas

Joshua D. Sugar, Farid El Gabaly, William Chueh, Kyle Fenton, Paul G. Kotula,
Velimir Radmilovic, Norman C. Bartelt, Joseph T. McKeown, Andreas M.
Glaeser, and Ron Gronsky



Sandia National Laboratories is a multi-program laboratory managed and operated by Sandia Corporation, a wholly owned subsidiary of Lockheed Martin Corporation, for the U.S. Department of Energy's National Nuclear Security Administration under contract DE-AC04-94AL85000.

Prof. Thomas' Legacy Influenced and Inspired My Education Greatly



- I was a graduate student of Andy Glaeser and Ron Gronsky from between 2001 and 2007
 - I would occasionally pass Prof. Thomas in halls when he was around and exchange friendly head nods
 - Students could be heard whispering “G.T. is here today” or “That’s G.T.”
- Ron Gronsky was an original Prof. Thomas student
- The capabilities and expertise that Prof. Thomas built for UC Berkeley materials science and at LBNL NCEM inspired the topic of my PhD dissertation work and continue to influence the research portfolio that I pursue today
 - Electron Microscopy
 - Metallurgy
 - Phase transformations

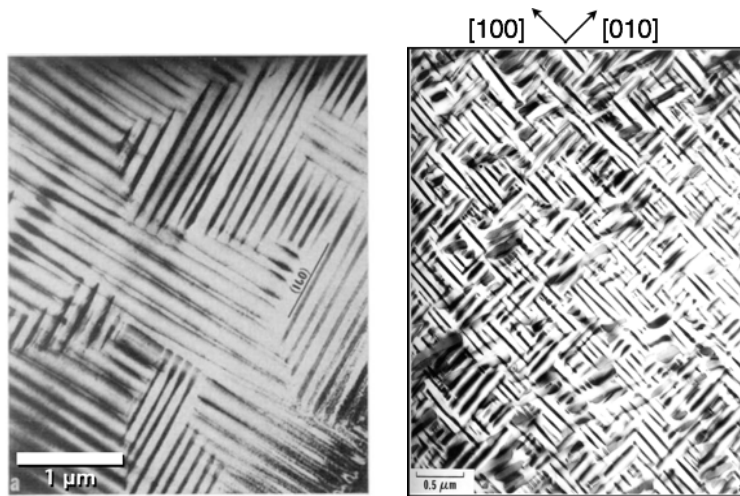
Prof. Thomas' 1970's Work Nucleated My PhD Dissertation Topic



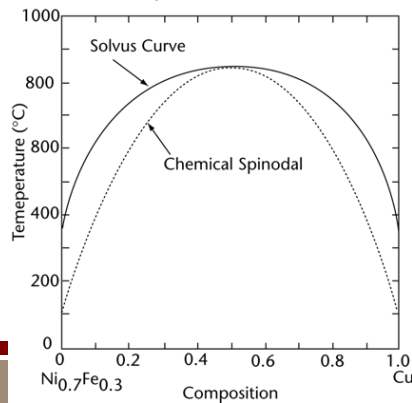
- Work from the 1970's incubated a number of my interests in materials science capabilities and problems
 - E.P. Butler and G. Thomas, *Structure and properties of spinodally decomposed Cu-Ni-Fe alloys*, Acta Met **18**, 1970.
 - R.J. Livak and G. Thomas, *Spinodally decomposed Cu-Ni-Fe alloys of asymmetrical compositions*, Acta Met **19**, 1971.
 - R.J. Livak and G. Thomas, *Loss of Coherency in spinodally decomposed Cu-Ni-Fe alloys*, Acta Met **22**, 1974.
 - R. Gronsby and G. Thomas, *Discontinuous coarsening of spinodally decomposed Cu-Ni-Fe alloys*, Acta Met **23**, 1975.
- This work inspired me to learn analytical TEM and continue to use this knowledge to solve materials problems

What was the next step for this work?

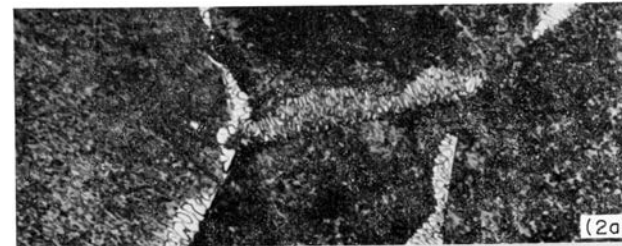
Can Decomposition of Cu-Ni-Fe Be Directed and Controlled Through Dimensional Confinement?



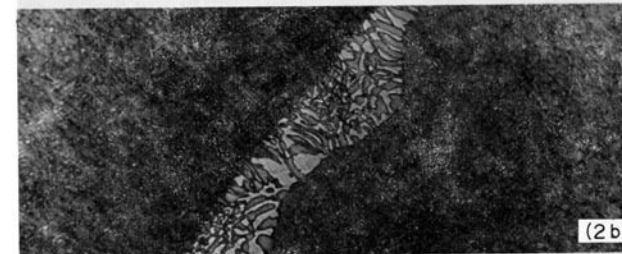
R.J. Livak and G. Thomas, *Spinodally decomposed Cu-Ni-Fe alloys of asymmetrical compositions*, Acta Met **19**, 1971.
R.J. Livak and G. Thomas, *Loss of Coherency in spinodally decomposed Cu-Ni-Fe alloys*, Acta Met **22**, 1974.



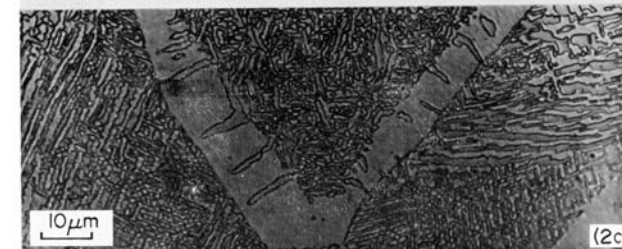
Clear development of modulated structure domains in bulk parallel to {100}



500 hr
654°C



500 hr
748°C

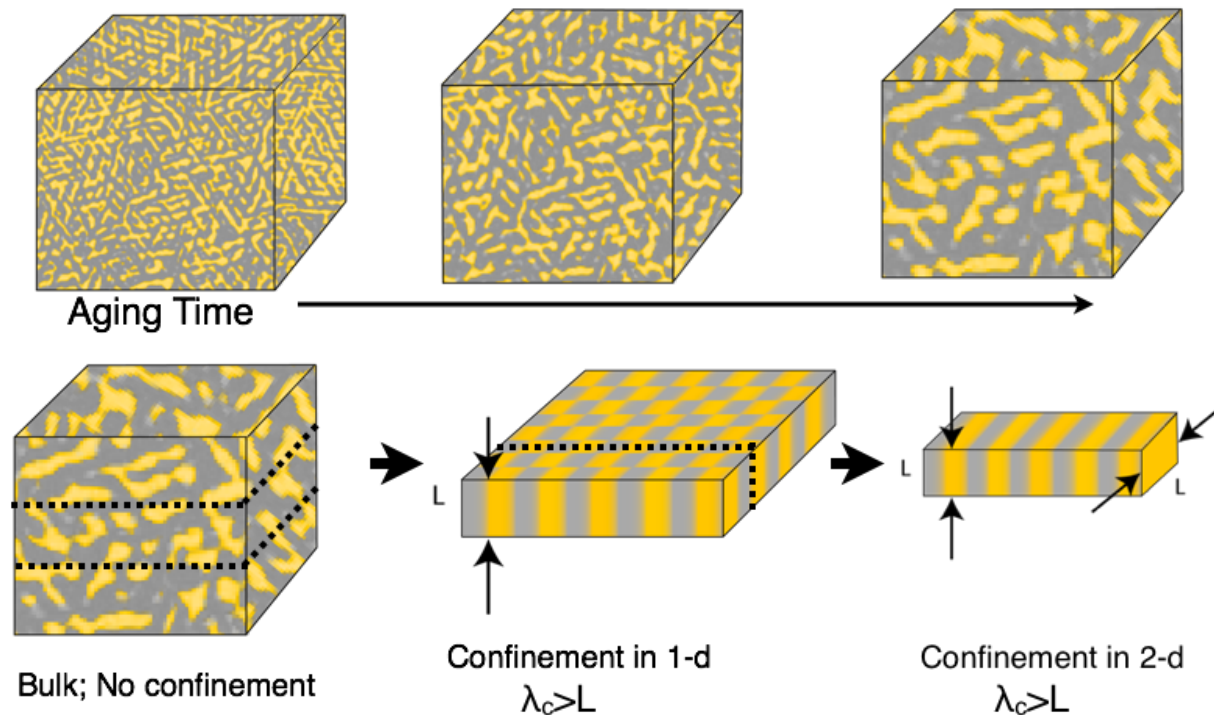


500 hr
838°C

R. Gronsky and G. Thomas, *Discontinuous coarsening of spinodally decomposed Cu-Ni-Fe alloys*, Acta Met **23**, 1975.

Coarse structure near grain boundaries suggests a single domain is possible in an extended volume

Can Decomposition of Cu-Ni-Fe Be Directed and Controlled Through Dimensional Confinement?



When spatial dimensions are smaller than preferred modulation wavelength, can wavelengths be frustrated such that modulation is directed in a specific direction?

Directing Confinement With Surface Faceting on $\{10\bar{1}0\}$ - Al_2O_3

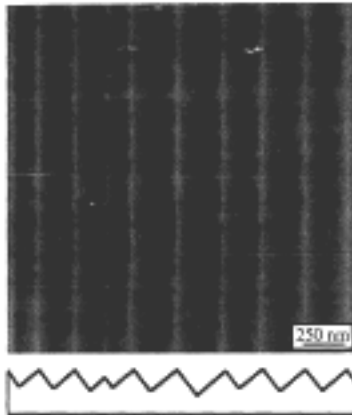


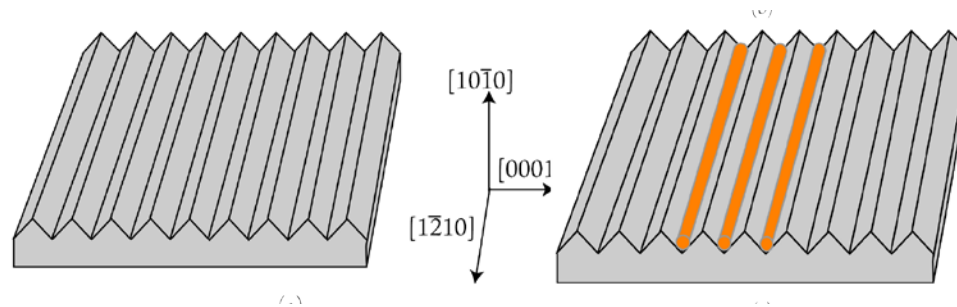
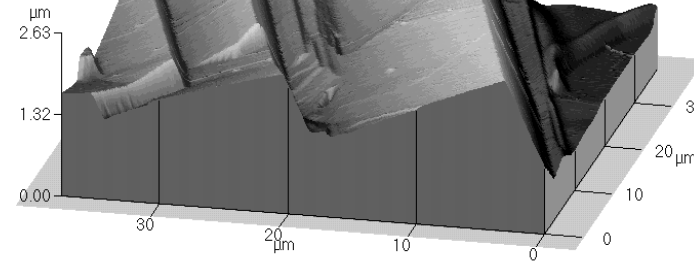
Fig. 1. Low-voltage (5 kV) secondary electron (SE) image of a $\{1000\}$ alumina surface that had been annealed for 6 h at 1430 °C. The schematic inset below the SE image shows a vertically corrugated profile of the hill-and-valley structure.

Heffelfinger, J.R., M.W. Bench, and C.B. Carter, *On the Faceting of Ceramic Surfaces*. Surface Science, 1995. **343**(1-2): p. L1161-L1166.

Can we direct film growth or dewetting to form long wires with a modulated structure?

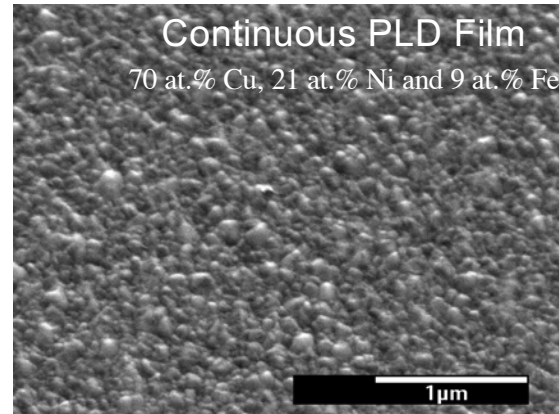
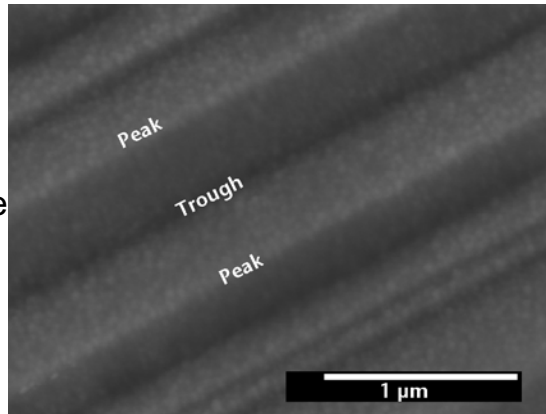
AFM of facets

24h
1700°C

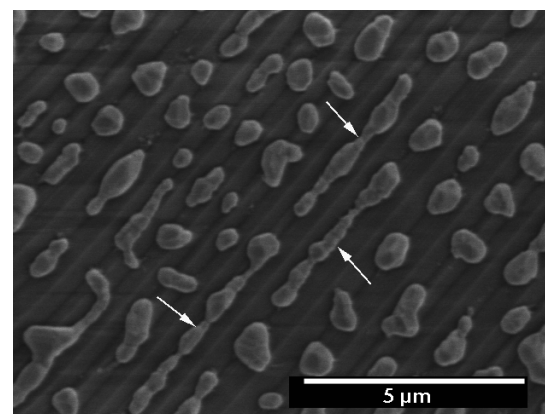
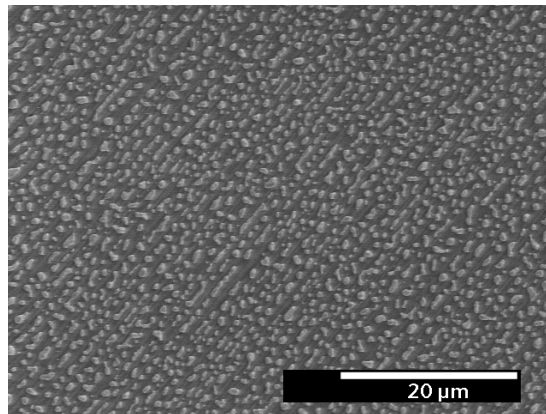


Directing Confinement With Surface Faceting on $\{10\bar{1}0\}$ - Al_2O_3

1600°C
12 h
~100 nm tall
~300 nm wide



700°C
8 h

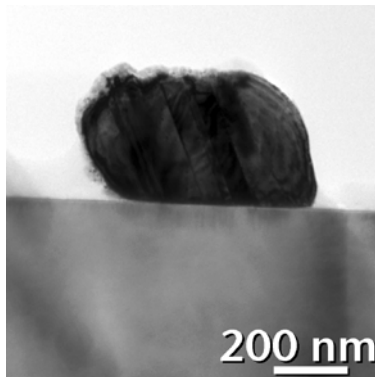
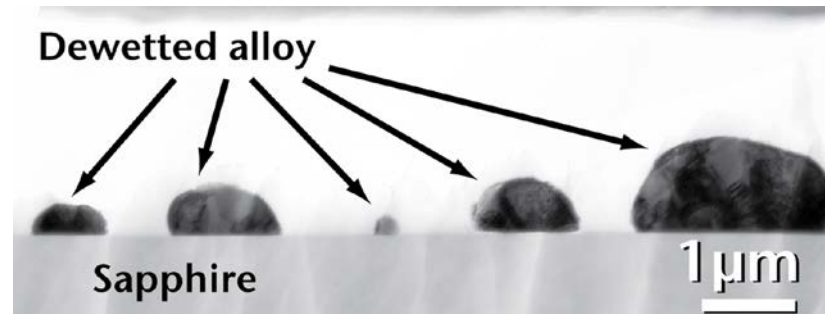


Dewetting Mechanisms

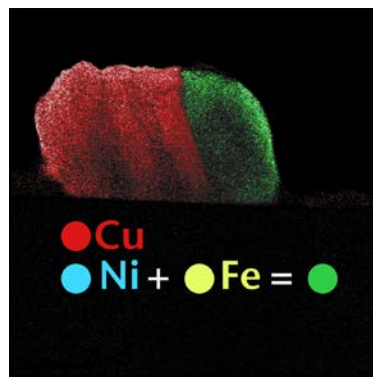
1. Lattice parameter difference between Cu and Ni-Fe rich phases causes thickness variation
 1. "spinodal dewetting"
 2. Bischof, J., et al., *Dewetting Modes of Thin Metallic Films: Nucleation of Holes and Spinodal Dewetting*. Physical Review Letters, 1996. **77**(Copyright (C) 2010 The American Physical Society): p. 1536.
2. Film surface curvature on a faceted surface leads to chemical potential-driven flow to troughs
 1. Basu, J., et al., *Nanopatterning by solid-state dewetting on reconstructed ceramic surfaces*. Applied Physics Letters, 2009. **94**(17).
3. Grain boundaries in polycrystalline film act as nucleation sites
 1. Genin, F.Y., W.W. Mullins, and P. Wynblatt, *Capillary instabilities in polycrystalline metallic foils – experimental observations of thermal pitting in nickel*. Acta Metallurgica Et Materialia, 1994. **42**(4): p. 1489-1492.

Directing Confinement With Surface Faceting on $\{10\bar{1}0\}$ - Al_2O_3

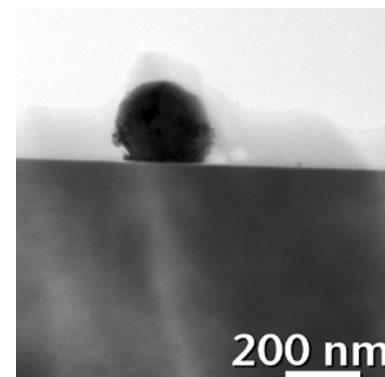
BF Image
20 h 700°C



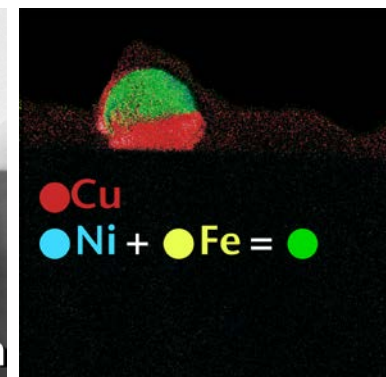
Zero loss Image



Composite ETEM map



Zero loss Image

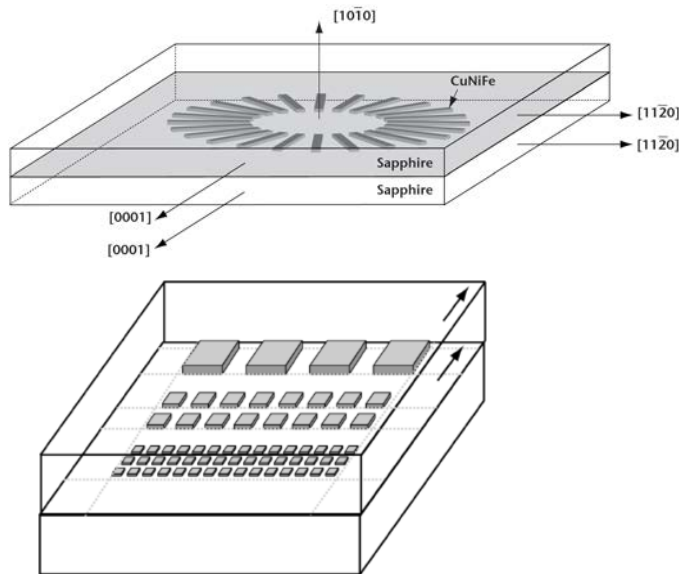


Composite EFTEM map

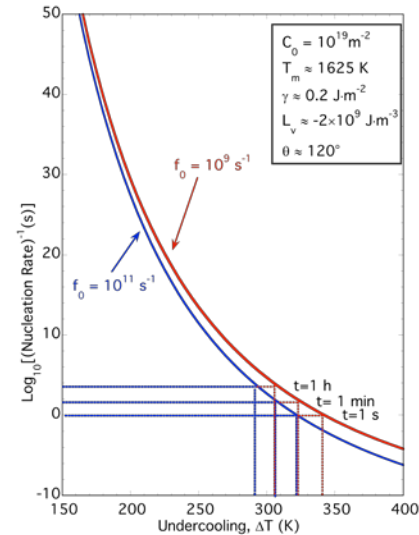
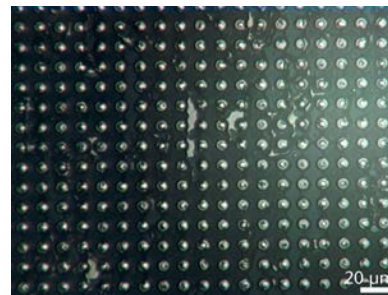
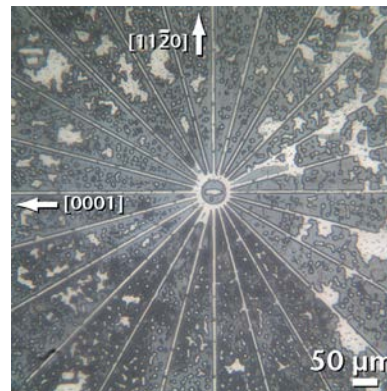
Alloy wires dewet into single bi-phase particles. What about full confinement to prevent dewetting?

Complete Encapsulation Induced Stabilization

1. Pattern definition with UV lithography
2. PLD of metallic alloy film
3. Diffusion bonding of sapphire crystals



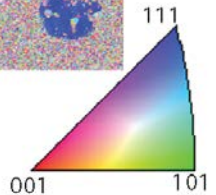
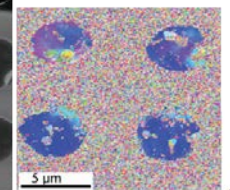
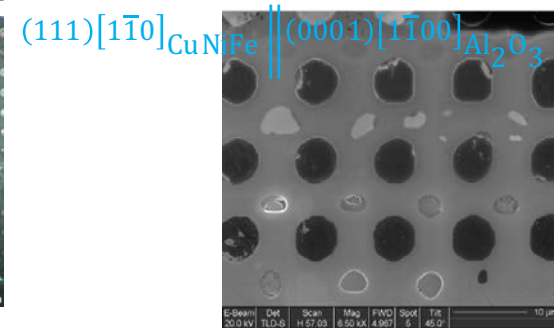
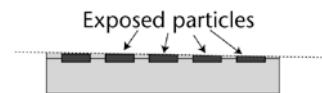
Remelt and solidify metal to form single crystals



Nucleation-Controlled Liquid Phase Epitaxy

- Cooled at a sufficiently slow cooling rate so that one nucleation even happen in each particle

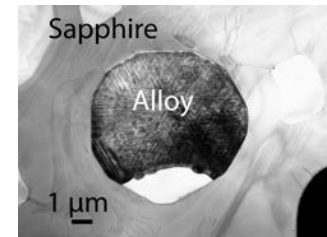
$$N = f_0 C_0 \exp\left(-\frac{16\pi\gamma_{sl}T_m^2}{3L_v k T(\Delta T)^2} S(\theta)\right)$$



Formation of Regular Pattern After Coarsening



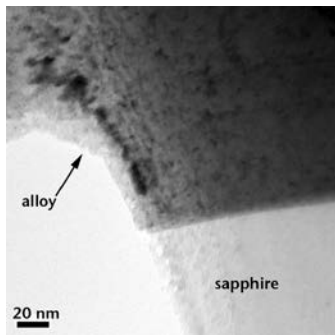
Cross sectional BF image



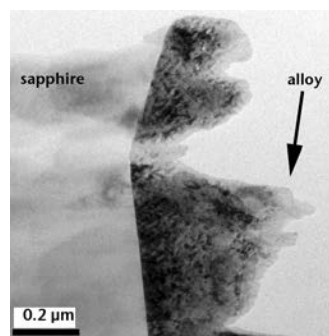
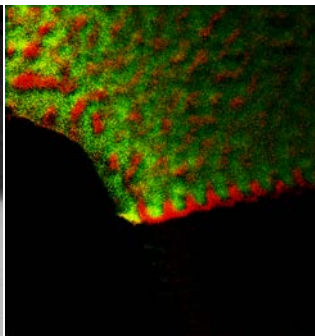
Top Down BF image



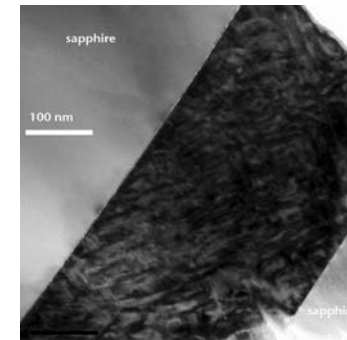
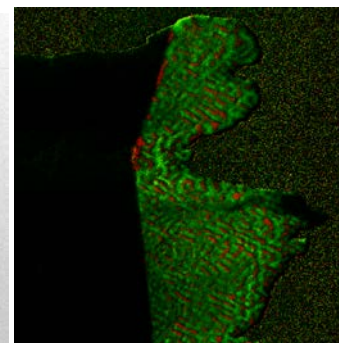
$$(111)[1\bar{1}0]_{\text{CuNiFe}} \parallel (0001)[1\bar{1}00]_{\text{Al}_2\text{O}_3}$$



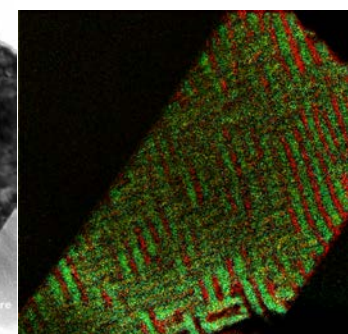
900°C Quench
 $\lambda \sim 15$ nm



700°C 30 min
 $\lambda \sim 21$ nm



700°C 90 min
 $\lambda \sim 23$ nm



Cu



Ni



Fe

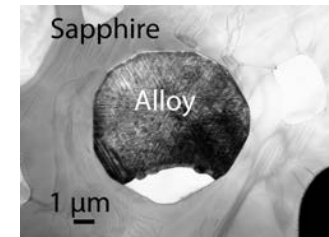


Ni and Fe Overlay

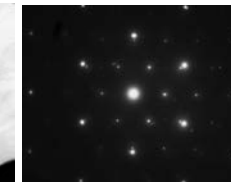
Formation of Regular Pattern After Coarsening



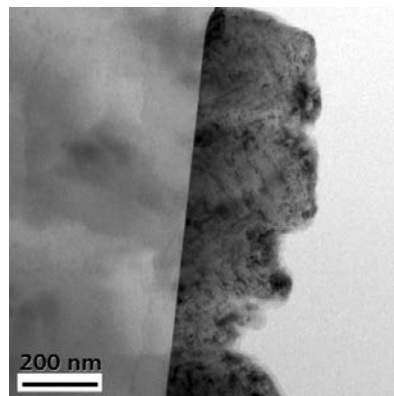
Cross sectional BF image



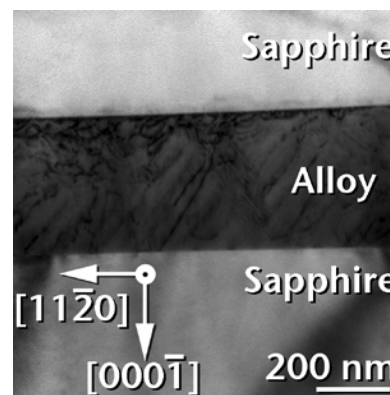
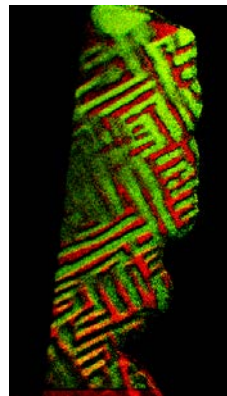
Top Down BF image



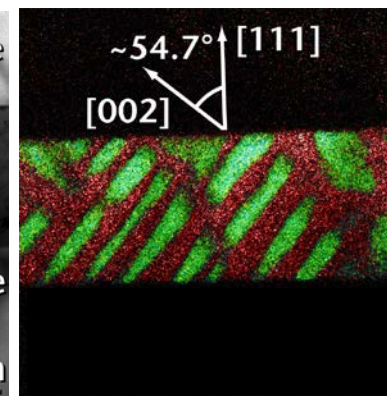
$$(111)[1\bar{1}0]_{\text{CuNiFe}} \parallel (0001)[1\bar{1}00]_{\text{Al}_2\text{O}_3}$$



700°C 300 min
 $\lambda \sim 40$ nm



700°C 11280 min
 $\lambda \sim 100$ nm



- Start to see one domain of modulated structure fill a cavity
- Orientation relationship does not allow for 90° patterning
- Initial stages of “directed decomposition” are visible at long aging times
- Probably a smaller cavity would aid in seeing this phenomena at earlier aging times



Cu



Ni



Fe

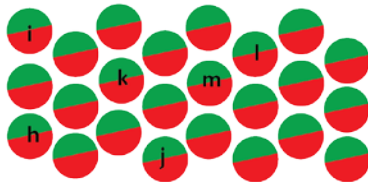


Ni and Fe Overlay

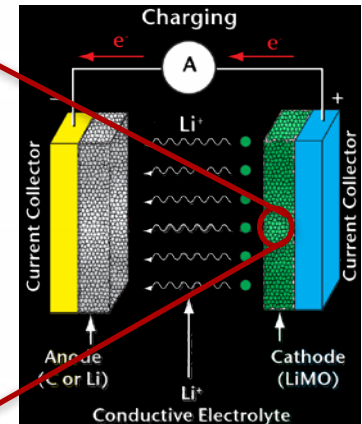
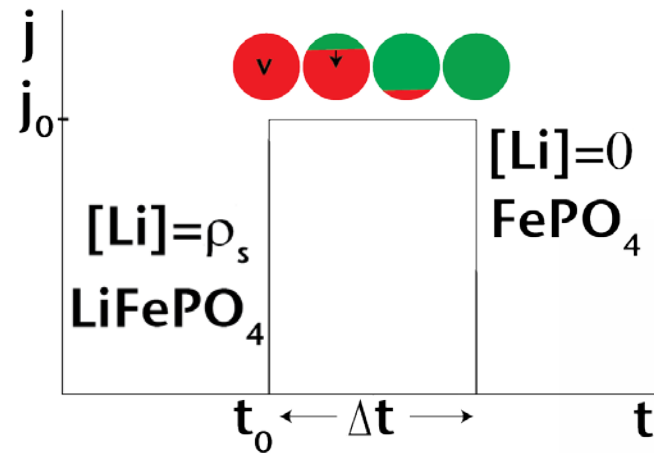
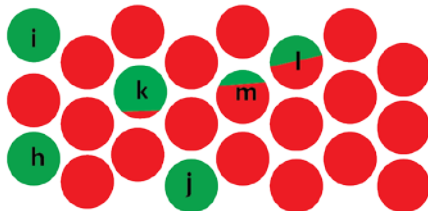
EFTEM Knowledge and Capability Inspired New Measurement for Li-ion Batteries

- Consider 2 Rate Limiting Cases for Charge/Discharge Transformation

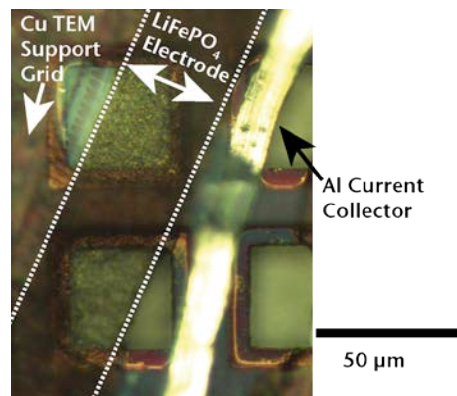
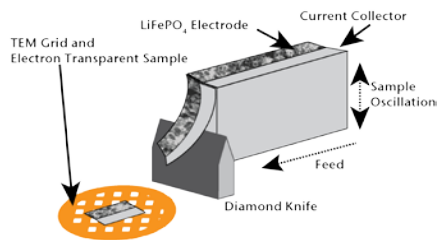
- Phase boundary migration rate limits



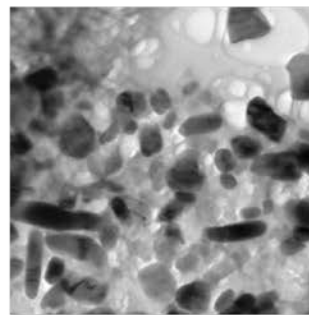
- Probability of new phase nucleating (nucleation rate) limits



EFTEM Enabled Measurement of Li Without Beam Damage Effects



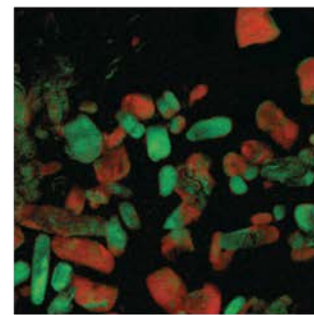
BF TEM image



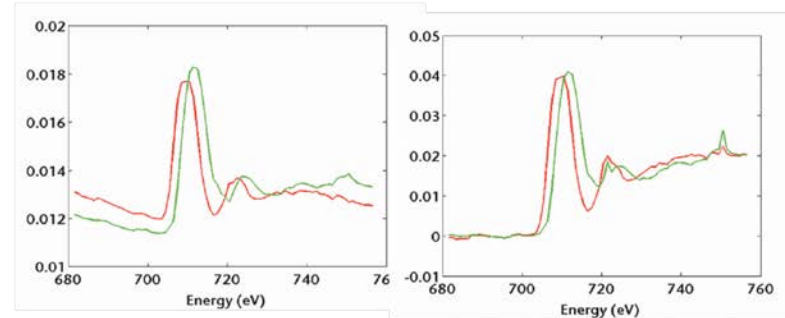
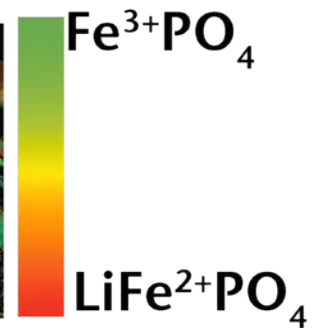
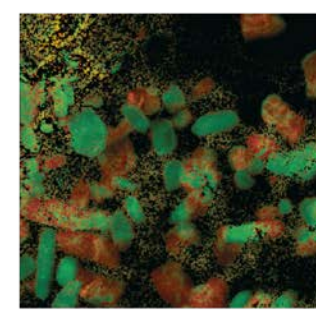
1 μm



EFTEM Map of Fe state from MSA pure components



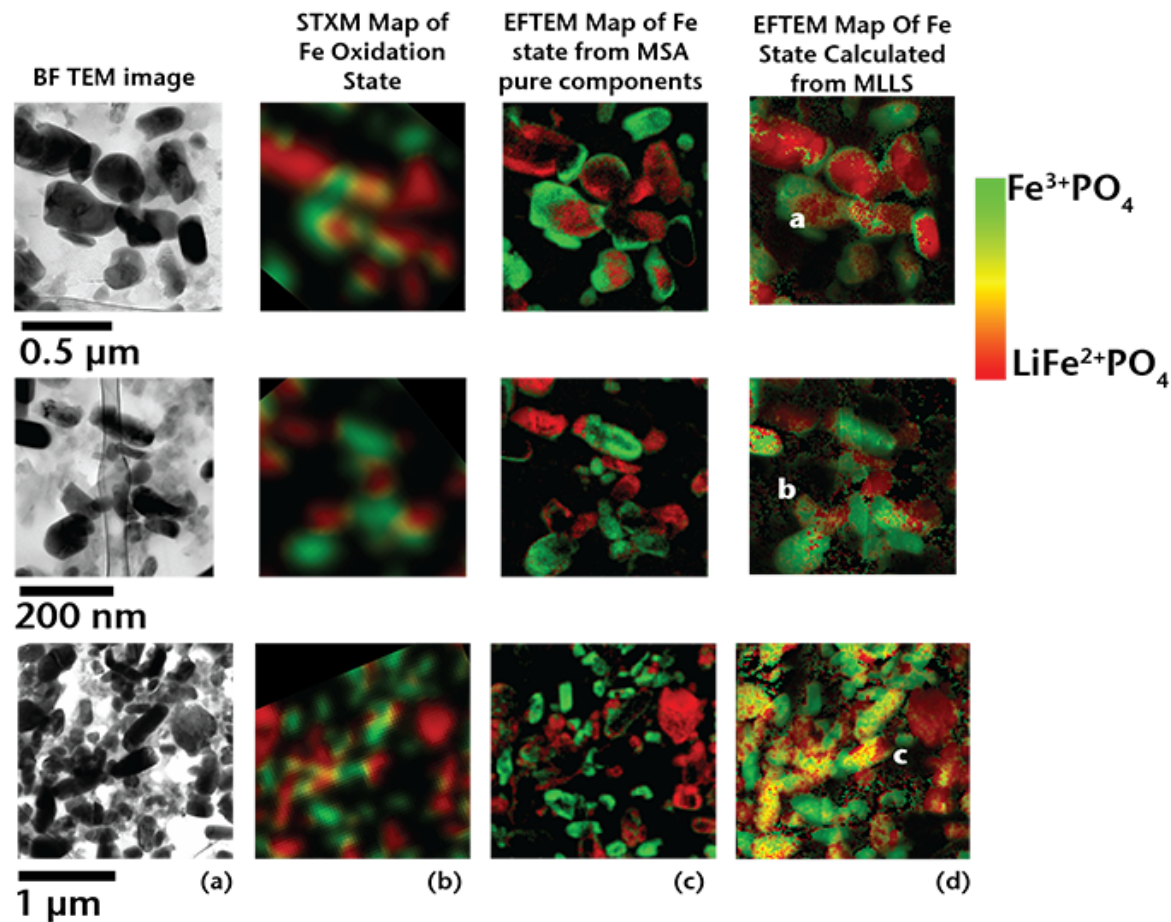
EFTEM Map Of Fe State Calculated from MLLS



LFP Cathode at 50% State of Charge

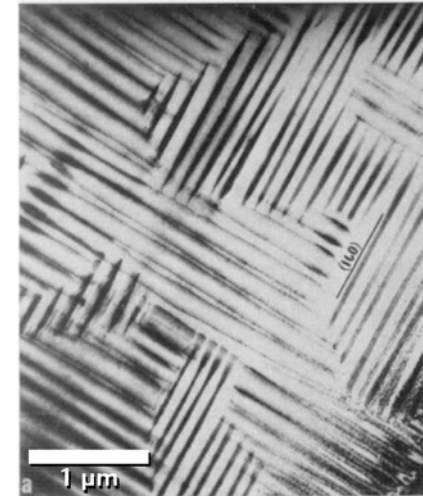
Confirm a Particle-By-Particle Transformation Reaction And Validate with STXM

- Most of the time particles are either completely LiFePO_4 or FePO_4
- A small fraction of particles appear mixed
 - Most of the time this is because particles overlapping in the projection view
 - Sometimes this is the result of non-linear thickness effects
 - Intensity is redistributed from the core edge energy as a result of plural scattering
 - Convolution of core-edge spectra with low-loss or plasmon spectrum
- Rarely do we find a single particle that exhibits both Fe^{2+} and Fe^{3+}
- Charging time is independent of time for individual particle and depends on nucleation rate for forming the new phase



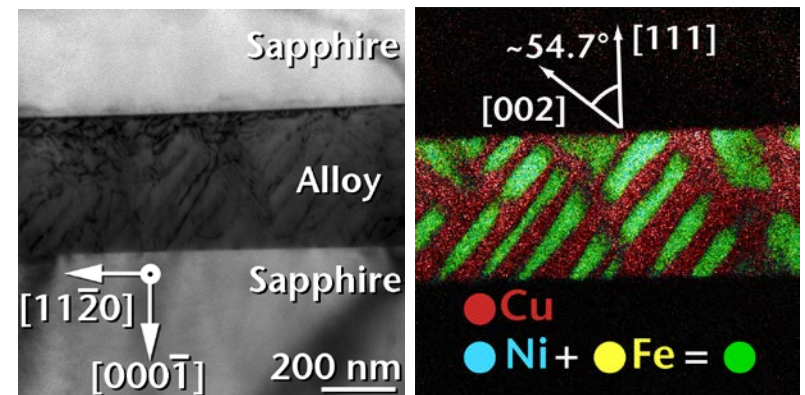
Summary and Conclusions

- Prof. Thomas work from the 1970's inspired the work that was the basis for my PhD dissertation work



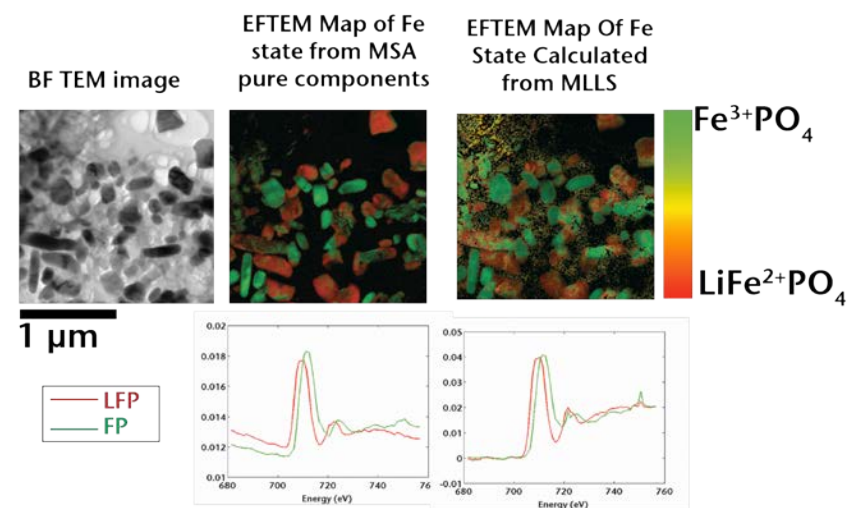
Summary and Conclusions

- Prof. Thomas work from the 1970's inspired the work that was the basis for my PhD dissertation work
- Ultimately my interest in understanding volumetric confinement in CuNiFe alloys required me to learn EFTEM, which began my interest and journey in analytical electron microscopy



Summary and Conclusions

- Prof. Thomas work from the 1970's inspired the work that was the basis for my PhD dissertation work
- Ultimately my interest in understanding volumetric confinement in CuNiFe alloys required me to learn EFTEM, which began my interest and journey in analytical electron microscopy
- I continue to use and develop the capabilities and knowledge that were inspired by Prof. Thomas to solve relevant materials problems today (Li-ion batteries)



Summary and Conclusions

- I spoke here today to share with you my personal experience of how I was inspired by Prof. Thomas and how his career impacted me even as a second generation student
- Prof. Thomas left a legacy that continues to impact and inspire microscopy and materials science professionals to solve hard problems and develop new capabilities
- I thank Prof. Thomas for the work he did at UC Berkeley and LBNL and am happy that I had the pleasure to benefit from it