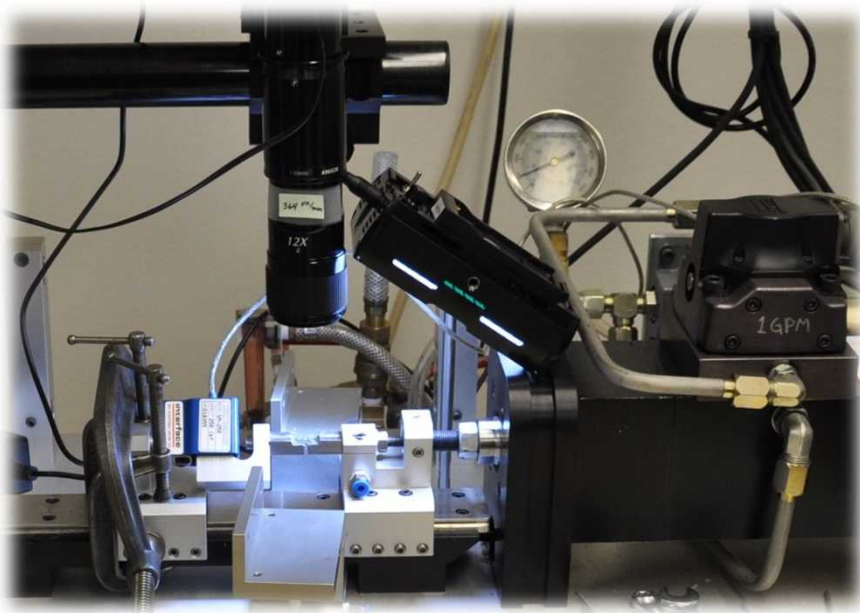
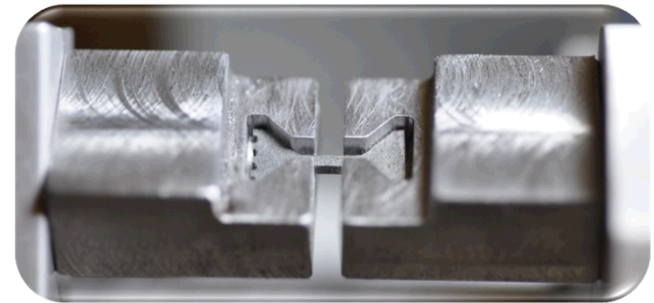


# Emergent homogenization techniques and effective dynamical properties

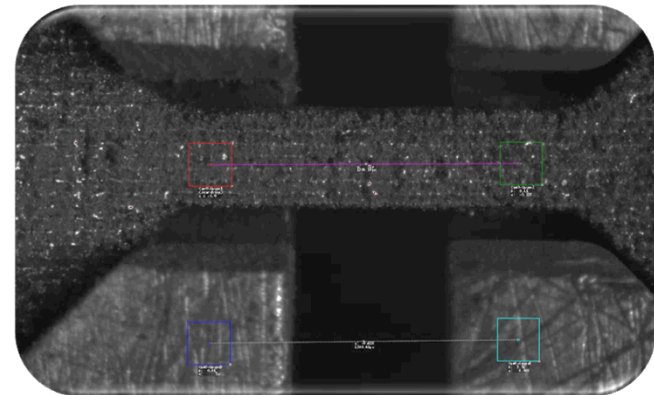
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tensile tester [7]



grippers w/1x1 mm gage section sample for testing [7]



Sample under test [7]

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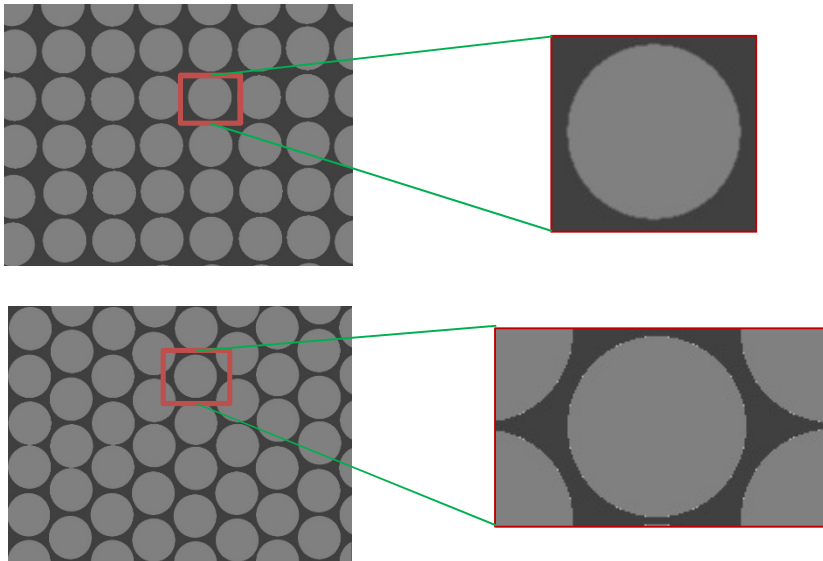
# Objective

Enable structure-property correlation analysis using a simplified microstructural representation:

- Generate statistically similar volume elements based on n-point correlation functions for heterogeneous material.
- Test the effect of different RVE configurations on the static and dynamic macroscale response using a quasi-explicit multiscale solver.

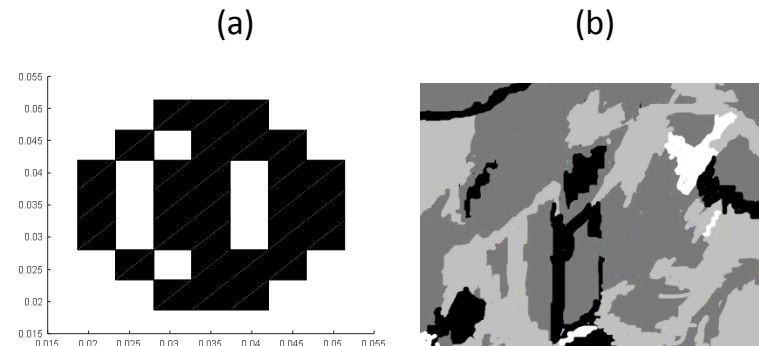
# ✧ Theory: Volume elements

- Representative volume element (RVE), is the smallest volume over which a measurement can be made that will yield a value that is representative of the whole.



**Fig. 1** Deduction of RVE for continuous fiber reinforced composite for square and hexagonal lattice

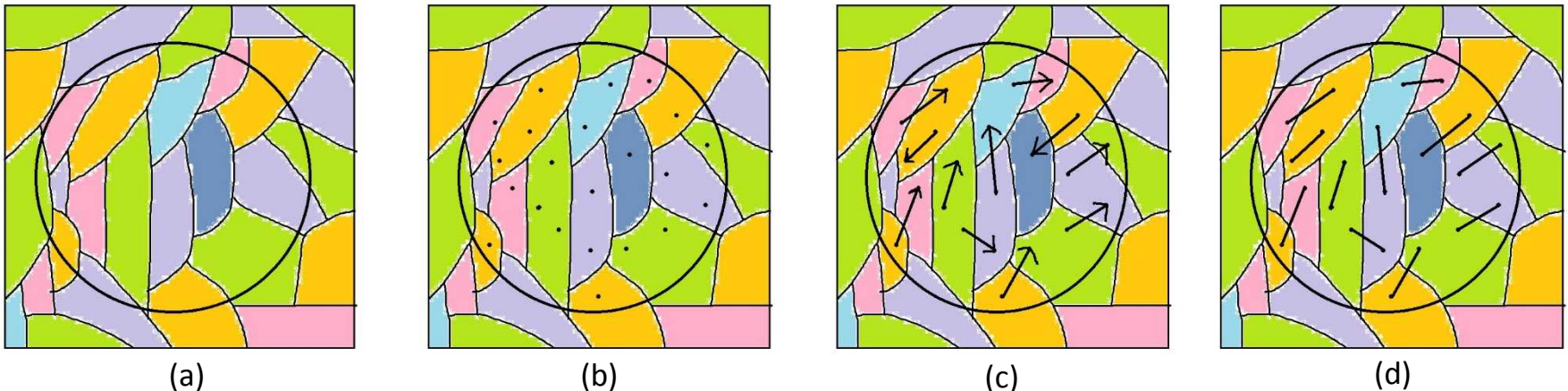
- Statistical volume element (SVE), also referred to as stochastic volume element in finite element analysis, takes into account statistics and variability in the microstructure of interest based on macroscopic property of interest.



**Fig. 2** Types of microstructure representation (a) Eigen microstructure – represented by discrete phases (here 0 & 1) (b) Non-eigen microstructure – represented by continuous phase

# ✧ Theory: Microstructural statistics

- **One-point correlation function:** probability that a randomly chosen point in the medium belongs to a phase
- **Two-point correlation function:** probability that two randomly chosen points  $\mathbf{x}_1$  and  $\mathbf{x}_2$  both lie in the same phase
- **Lineal-path function:** probability that the entire line segment between points  $\mathbf{x}_1$  and  $\mathbf{x}_2$  lies in the same phase
- **Two-point cluster function:** the probability that two randomly chosen points  $\mathbf{x}_1$  and  $\mathbf{x}_2$  belong to the same cluster of a phase

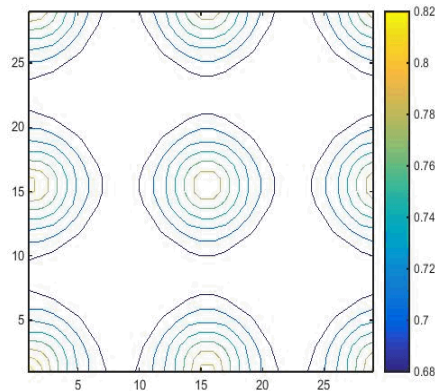
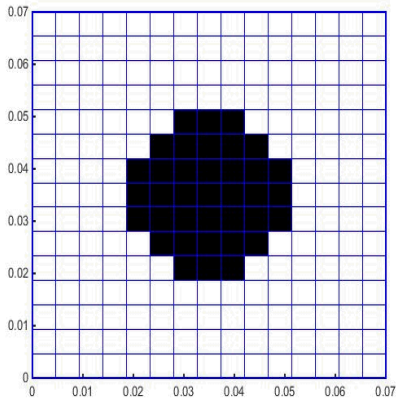


**Fig. 3** Depiction of microstructure statistics (a) given microstructure (b) One point correlation (c) Two point correlation (d) Lineal-path correlation

# ✧ Microstructure reconstruction algorithm

## Phase recovery iterative algorithm

**Step 1:** Deduction of statistical parameter – two point correlation



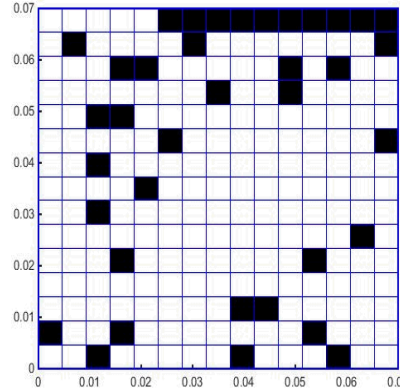
### Target microstructure

- $\phi_1$  (black phase) = 0.16
- $I$  : target indicator function
- $N \times N$  : grid size

### Target 2-point correlation

$$= \frac{1}{N^2} \text{fft}^{-1} \left[ (\text{fft}(I) * \text{fft}(I)) \right]$$

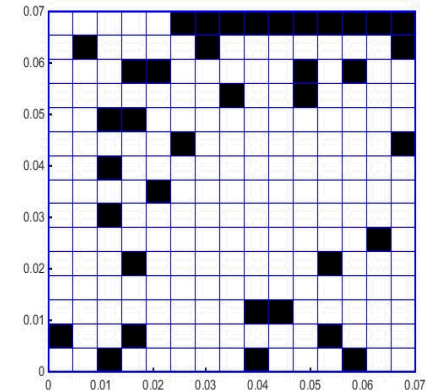
**Step 2:** Initialize optimization



### Gaussian random microstructure as initial iteration

- $\phi_1 = 0.16$  (constraint)

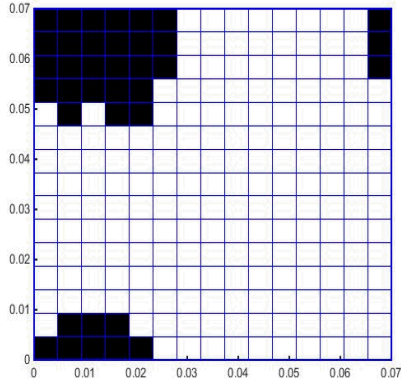
**Step 3:** Phase recovery iteration



### Iteration steps

- Fourier amp. & angle
- Replace amp. with target amplitude
- Assign 0's & 1's based on maximum likelihood

**Step 4: Invoking periodic structure constraint**

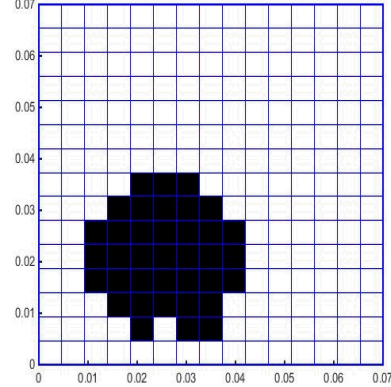


**Optimized microstructure**

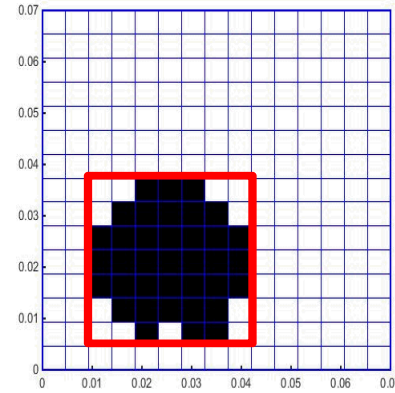
- $\varphi_1 = 0.16$
- Error

$$\triangleq \left\| \text{Target-Recon} \right\|_{\text{F}} = 0.032$$

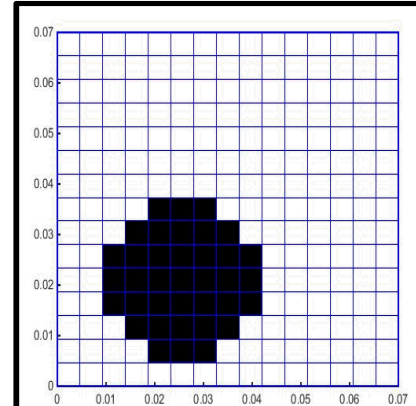
**Step 5: Selection of cluster of particles**



**Reconstructed with  
periodic boundary  
condition**



**Optimization is  
performed only on the  
selected cluster through  
Simulated Annealing**

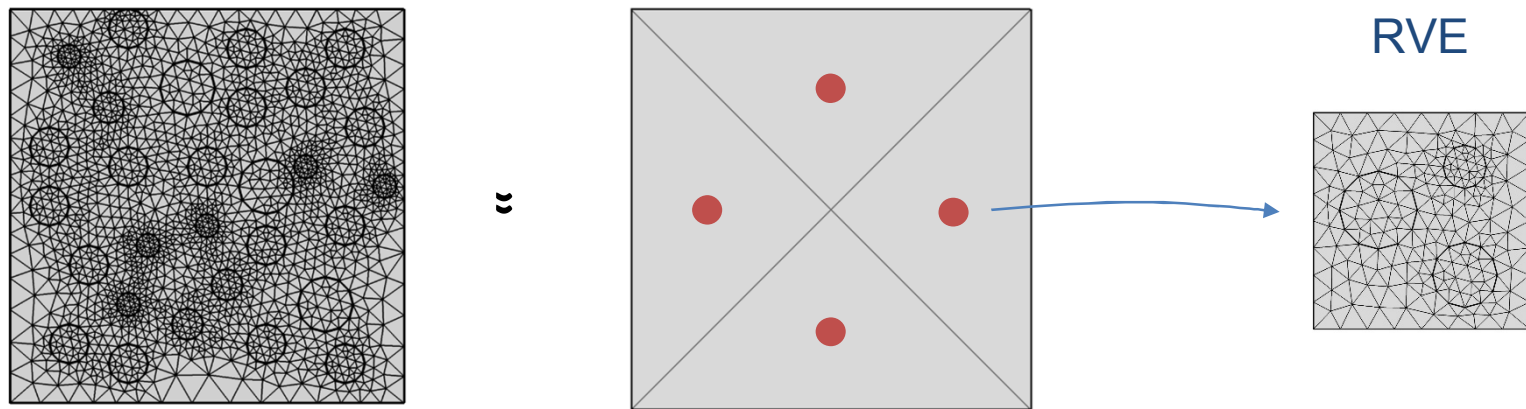


**Final reconstructed  
microstructure**



# ✧ Theory: FE<sup>2</sup> method

- Finite element squared (FE<sup>2</sup>) method is a computational homogenization approach, which is based on the solution of two boundary value problems, one for the macroscopic and one for the RVE scale.



**Fig. 4** Schematic figure of single scale and multiscale finite element mesh.

- A multiscale solver is recently formulated in the spirit of FE<sup>2</sup> and is capable to deal with general loading conditions, including inertia effects. Each time incremental problem can be solved exactly with a single Newton-Raphson iteration with a constant Hessian, representing a quasi-explicit multiscale solver (QEMS).



# ✧ Quasi-explicit multiscale solver

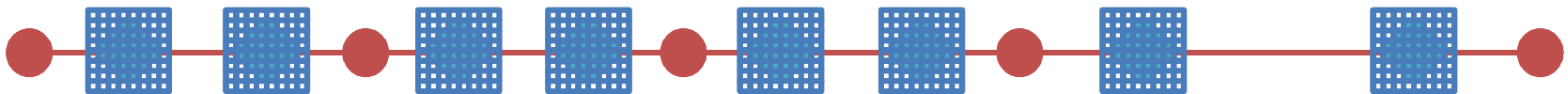
**Step-0** Consider a 1-D macroscopic mesh with  $n$  elements and  $n+1$  nodes.



**Step-1** Setup initial configuration (two RVEs per 1-D element).



**Step-2** Apply increment to the macroscopic boundary conditions.  
Get boundary condition for all RVEs and solve RVE as BVP.



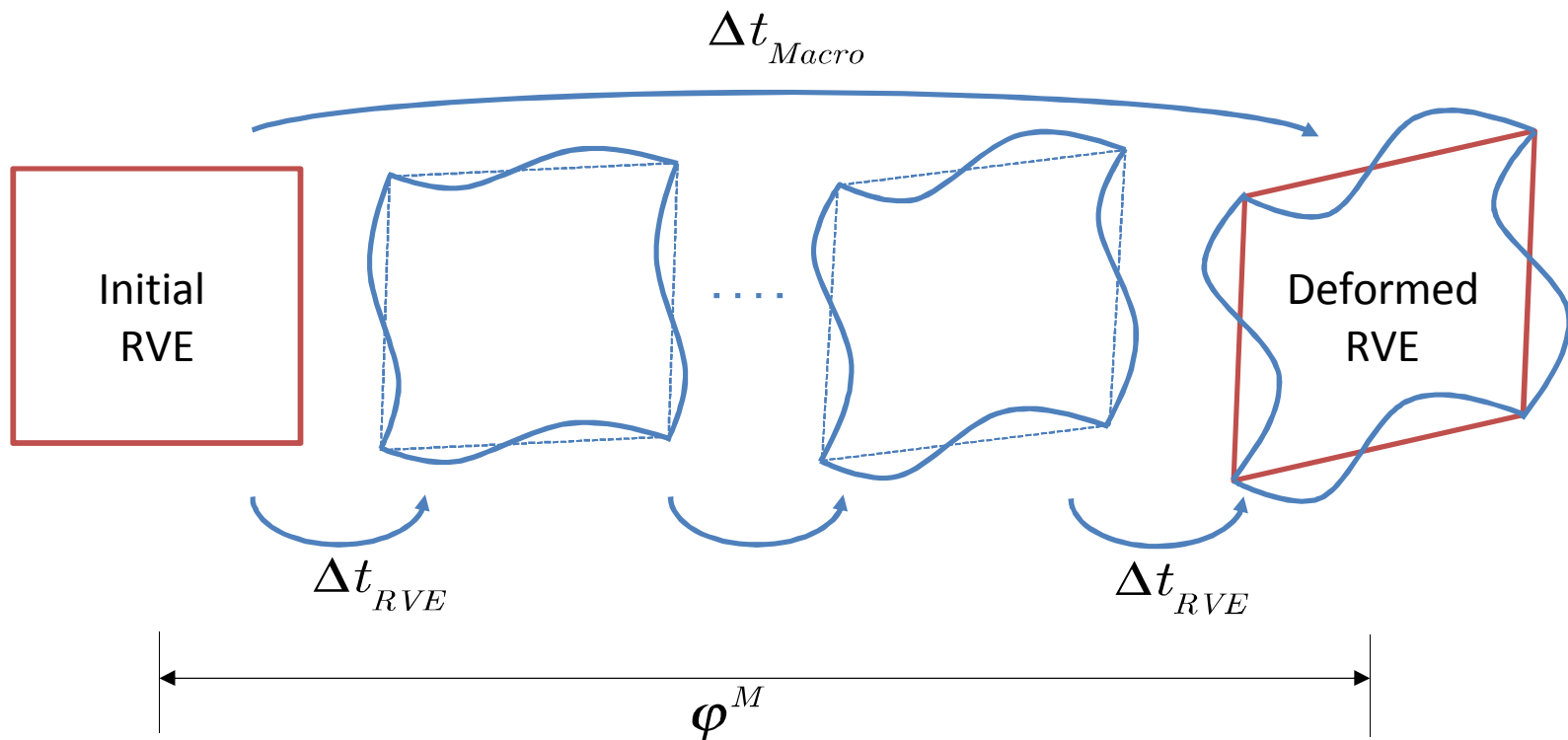
**Step-3** Compute macroscopic force and update macroscopic nodes.



**Step-4** Solve all RVE problems in the current macroscopic configuration.  
Update information for the next time step.

# ✧ Improvement of QEMS

- The original multiscale solver uses explicit Newmark method, which is conditionally stable. Instead of using same time step, different time steps could be used in the multiscale solver. In the step 4 of the flow chart, the boundary condition for all the RVE could be applied adaptively.

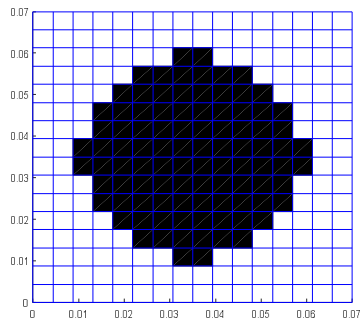


**Fig. 5** Schematic figure of sub-cycling in multiscale solver.

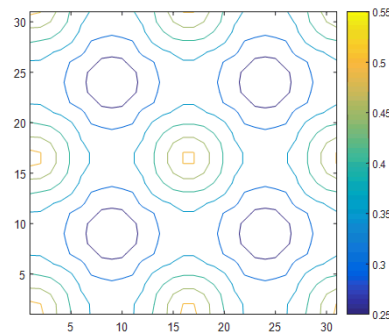
# ✧ Case Study

- Problem Description

A microstructure with a circular inclusion is studied with both a 16X16 and a 8X8 pixel grid. These microstructures represent actual and pixelated volume element image. The coarser mesh microstructure is generated iteratively by minimizing difference in two point correlation with the volume fraction constraint. QEMS is used to test the macroscopic response of the two sets of RVEs.

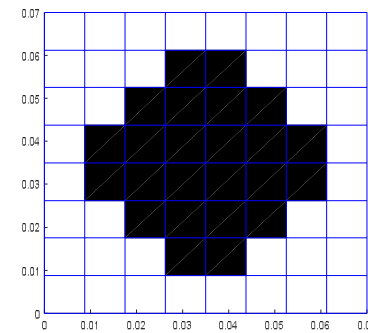


(a)

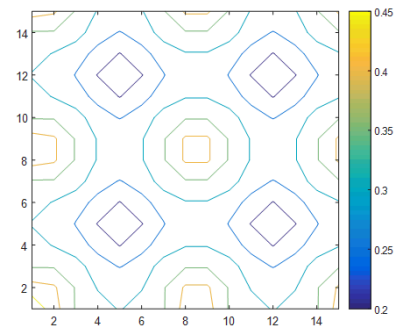


(b)

**Fig. 6** (a) Microstructure with circular inclusion on 16X16 pixel grid.  
(b) Two point correlation of microstructure.

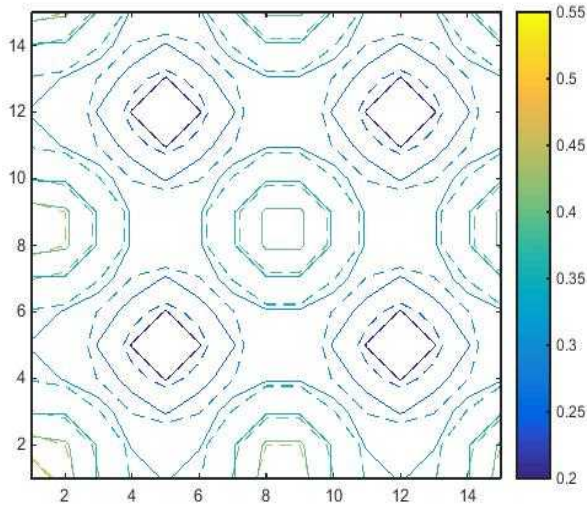


(a)



(b)

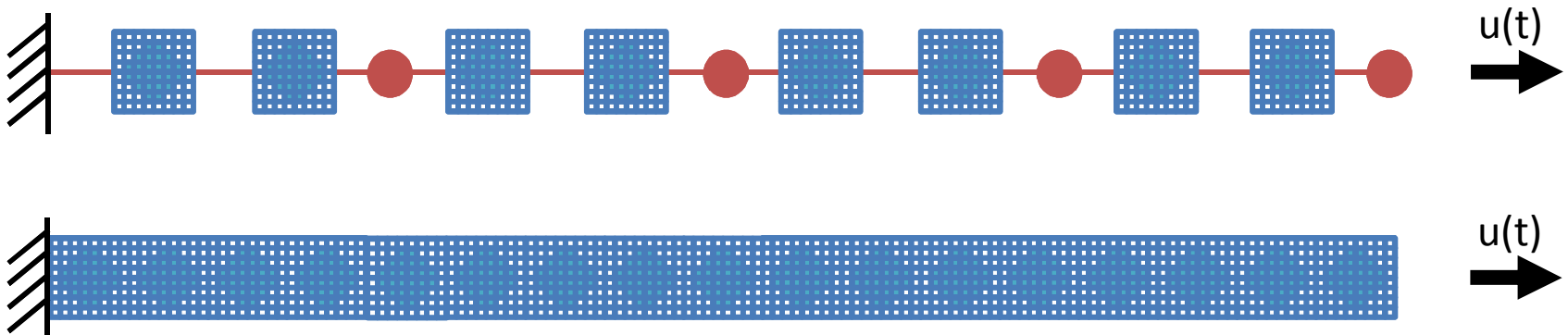
**Fig. 7** (a) Microstructure with circular inclusion on 8X8 pixel grid.  
(b) Two point correlation of microstructure.



**Fig. 8** Comparison of two point correlation deduced from original 8X8 pixel grid (-) and interpolated from 16X16 (--) pixel grid.

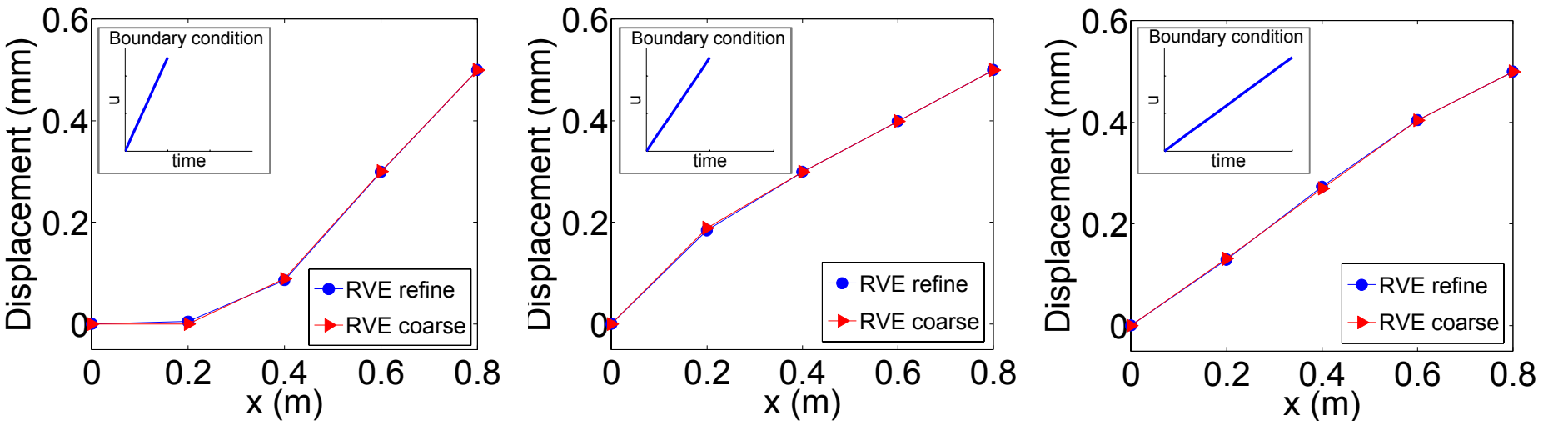
- Comparison of original and interpolated correlation shows the approximation during the mesh coarsening
- Interpolated values are higher than original data in general
- Two point correlation for  $r=0$  (or 1 point correlation) are close for both cases
- Direct interpolation can cause distortion in complex asymmetric volume element

## • Computational results



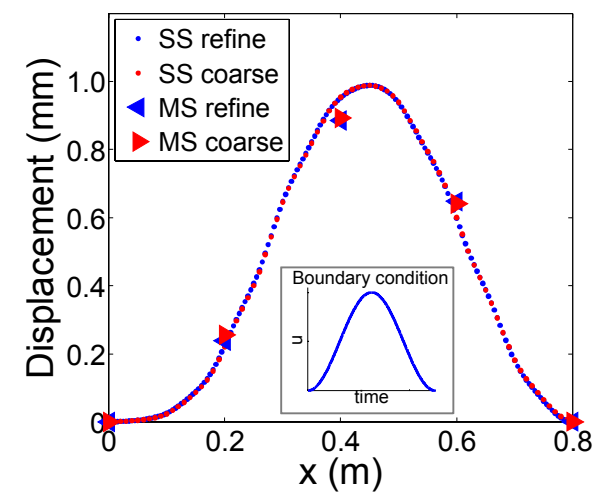
**Fig. 9** Multiscale and single scale finite element model.

## Quasi-static response by multiscale solver (MS)

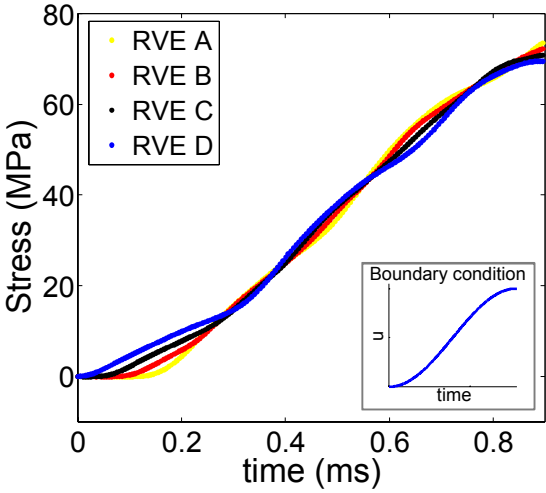


**Fig. 10** Displacement distribution along macroscopic domain at final time with different loading rate.

## Dynamic response and effective Young’s modulus



**Fig. 11** Displacement distribution along the macroscopic domain at final time.



**Fig. 12** Time evolution of the volume average of stress along the macro domain.

Materials	E (GPa)	density (kg/m³)
Inclusion	20	1000
Matrix (steel)	200	7000
Effective $\rho(\lambda / T)^2$	114.9	4750
Effective $\sigma / \varepsilon$	116.8	4750

**Table. 1** Material properties of the composite and the calculated Young’s modulus.

# Conclusions

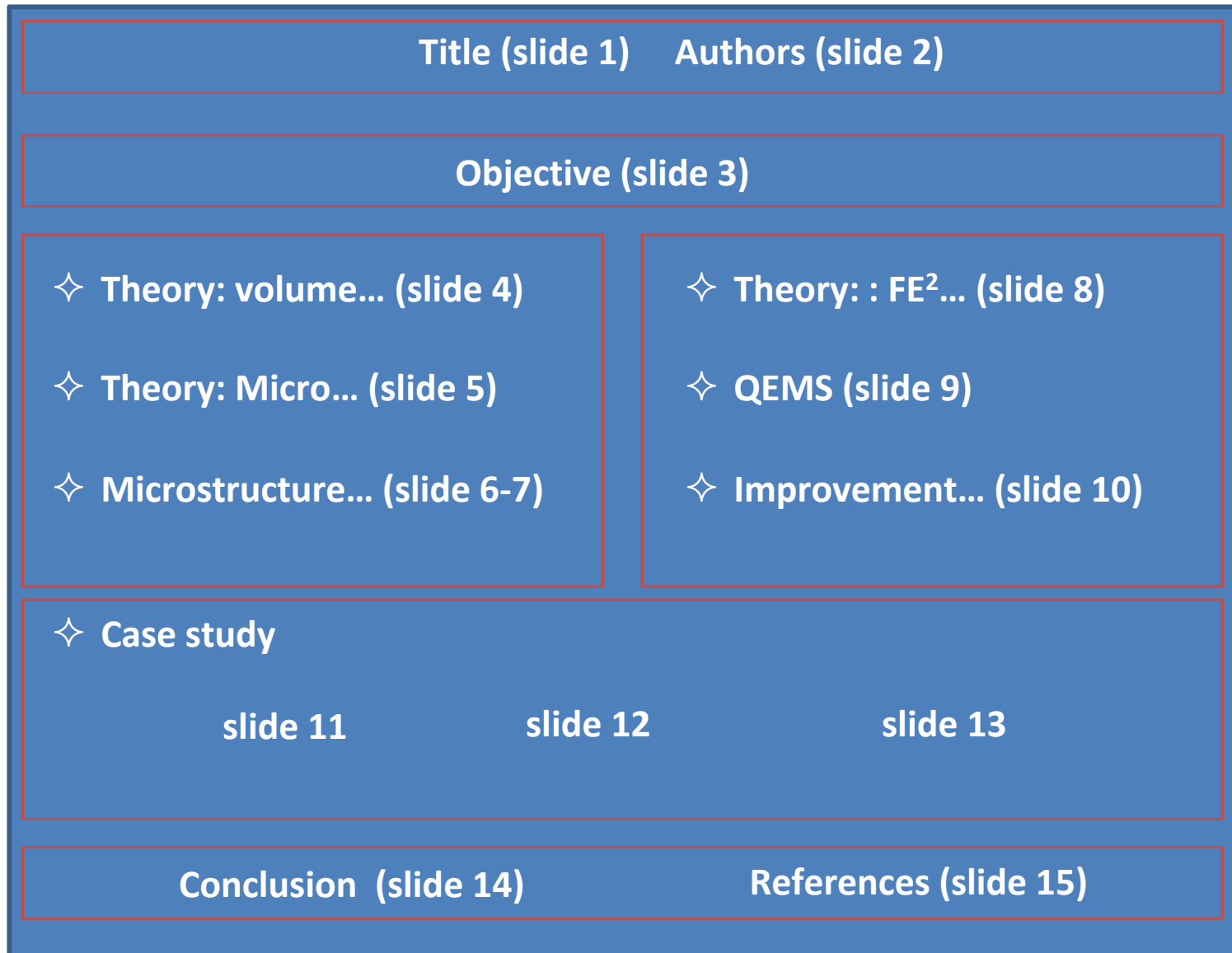
- Using the 2-point correlation function via fft, along with simulated annealing, a microstructure can be recreated with various grid sizes.
- With the coarsening procedure, while keeping the similar statistic property of RVEs, the presented macroscopic displacement distribution is similar.
- Effective Young's modulus computed by the wavelength and evolution of static stress are shown to be in good agreement.

# References

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2. Cecen A., Fast T., Kalidindi S.R., Versatile algorithms for the computation of 2-point spatial correlations in quantifying material structure, Integrating Materials and Manufacturing Innovation 5:1, 2016
3. Fullwood D.T., Niezgodka S.R., Kalidindi S.R., Microstructure reconstructions from 2-point statistics using phase-recovery algorithms, Acta Materialia, 56, 942–948, 2008
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5. Liu, C. and Reina C., Computational homogenization of heterogeneous media under dynamic loading, arXiv preprint arXiv: 1510.02310.
6. Pham, K., Kouznetsova, V., Geers, M., Transient computational homogenization for heterogeneous materials under dynamic excitation. Journal of the Mechanics and Physics of Solids, 2013, 61 (11), 2125-2146.
7. The credit of the photo of experimental instruments belongs to Brad Boyce and Brad Salzbrenner of Sandia National Labs.



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# Another Suggested Layout of slides:

