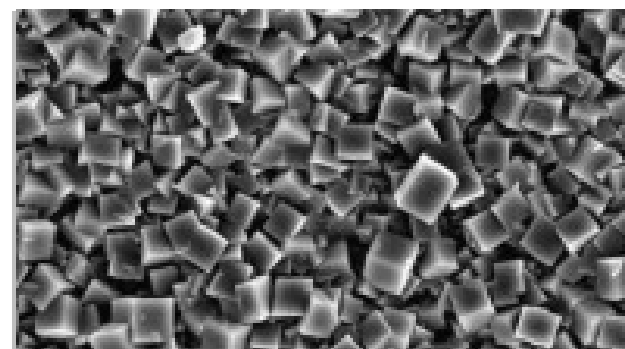
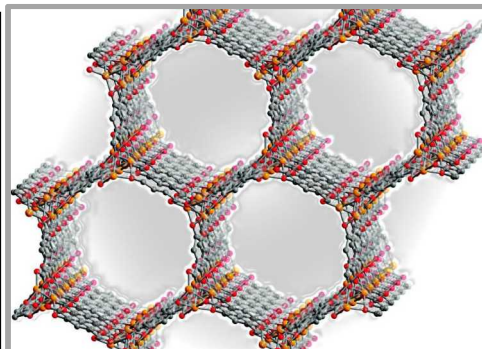


Exceptional service in the national interest



DFT Modeling of Metal-Organic Frameworks for Oxygen-Nitrogen Separation

Tina M. Nenoff,* Marie V. Parkes, Jeffery A. Greathouse, David Hart,
Dorina Sava Gallis

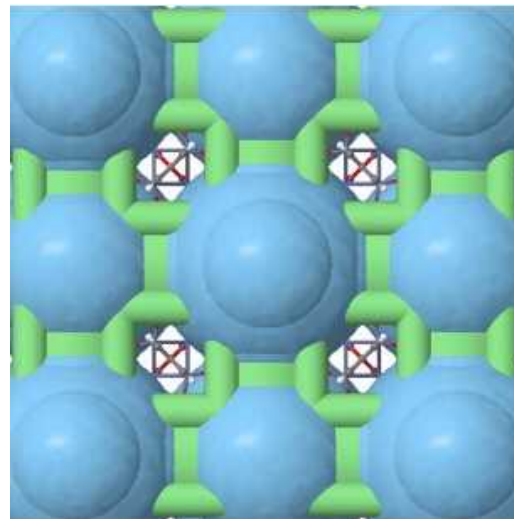
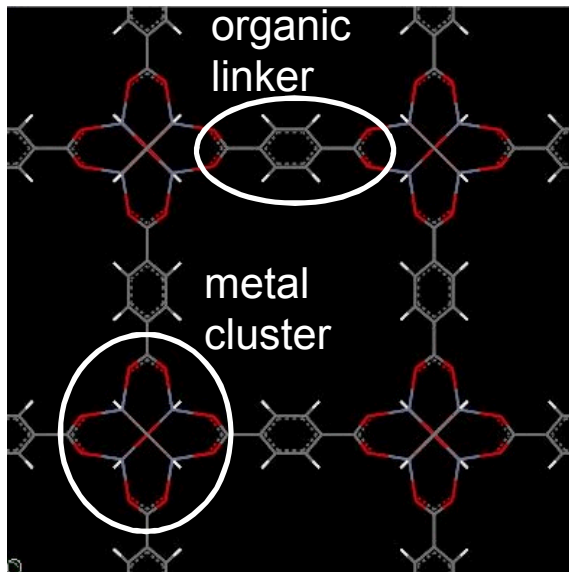
Sandia National Laboratories, Albuquerque, NM USA



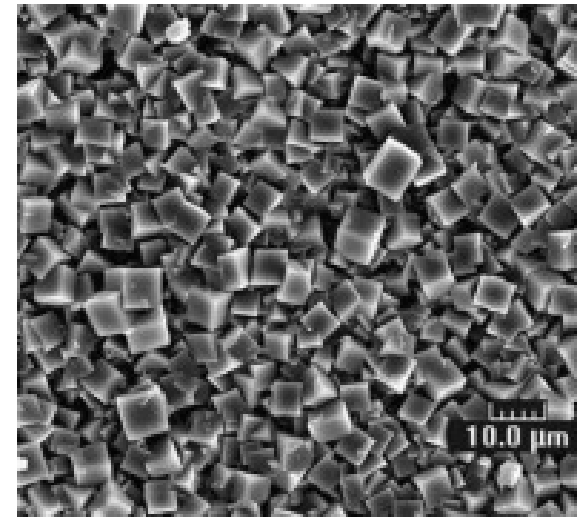
Funded by the Laboratory Directed Research and Development (LDRD) Program. Sandia National Laboratories is a multi-program laboratory managed and operated by Sandia Corporation, a wholly owned subsidiary of Lockheed Martin Corporation, for the U.S. Department of Energy's National Nuclear Security Administration under contract DE-AC04-94AL85000.

Metal-Organic Frameworks (MOFs)

- Porous solids made of an extended network of metal clusters coordinated to multidentate organic linkers
- Surface areas up to $6000 \text{ m}^2\text{g}^{-1}$
- “Crystalline sponges”



MMM, 2013, 165, 32-39

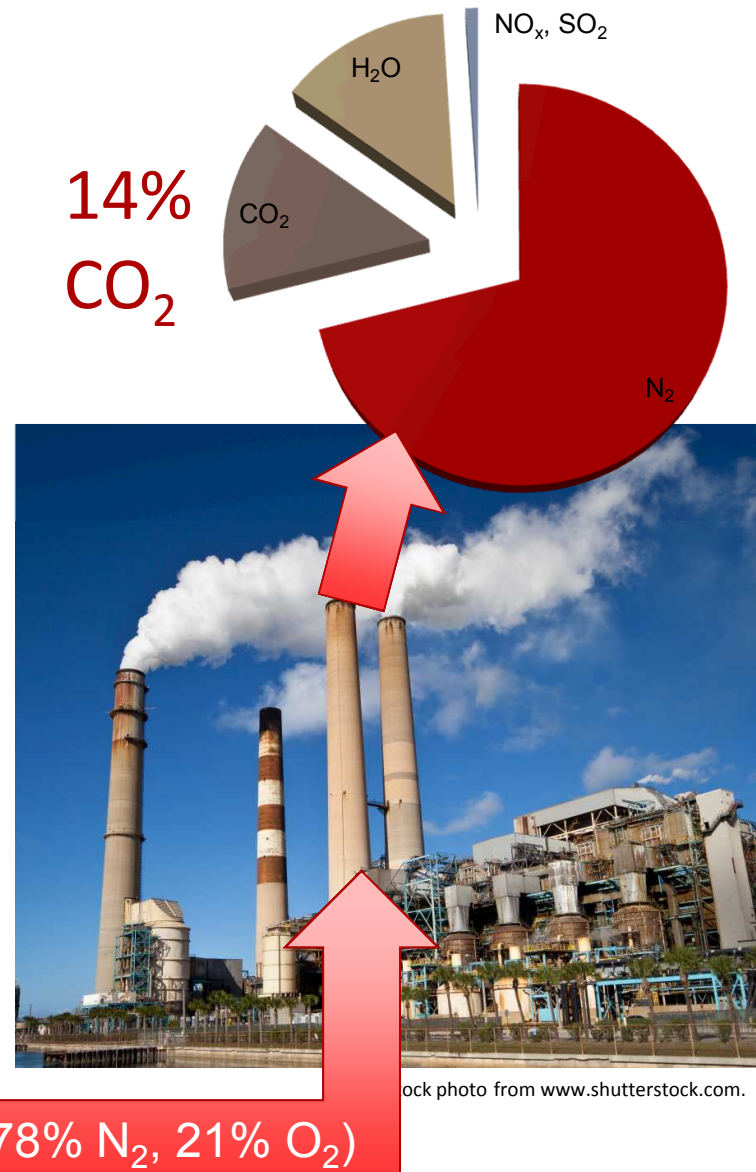


MMM, 2009, 123, 100-106.

Oxyfuel for Cleaner Power Plants

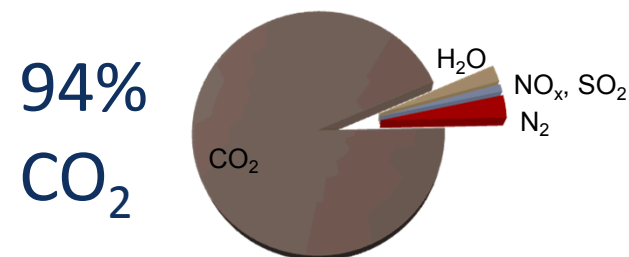
- Coal-burning power plants major source of carbon dioxide emissions
- Interest in capturing CO₂ emissions

What comes out depends on what goes in!

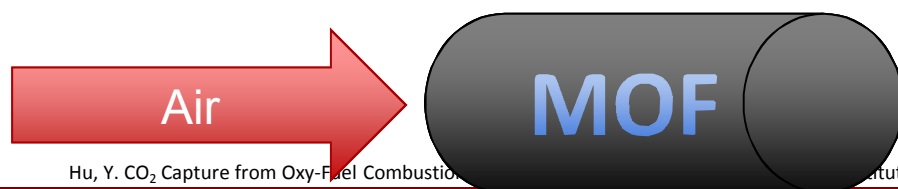


Oxyfuel for Cleaner Power Plants

- Oxyfuel advantages
 - 75% reduction in flue gas volume
 - Decreased NO_x emissions
 - Greater thermal efficiency
 - More energy efficient (not using energy to heat N_2)
 - Easier to capture CO_2
- Goal: MOFs for O_2 separation
 - Alternative to energy-intensive cryogenic separation



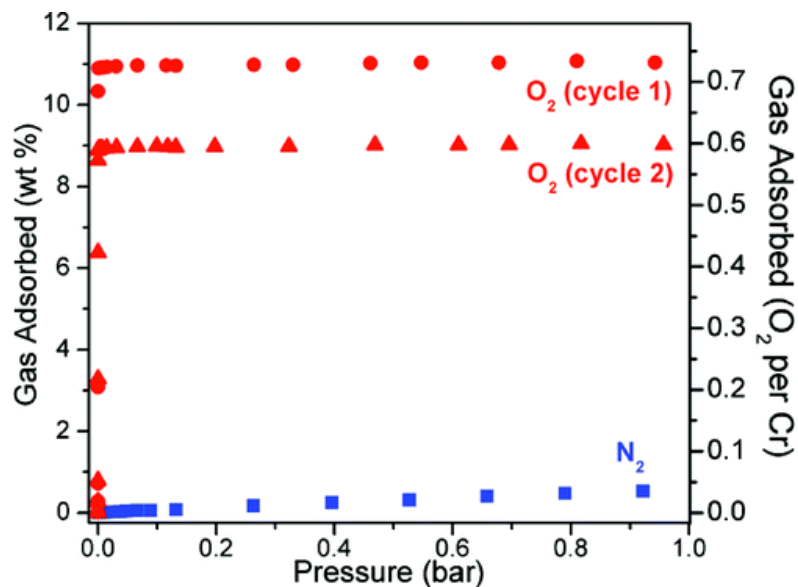
Stock photo from www.shutterstock.com.



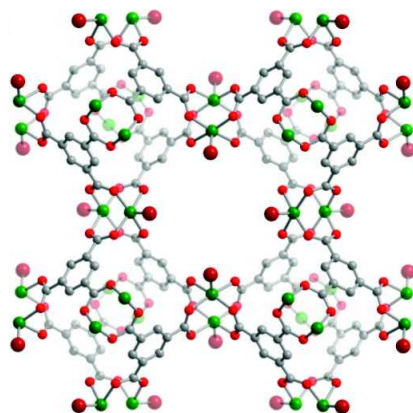
Hu, Y. CO_2 Capture from Oxy-Fuel Combustion

Institute of Technology, Stockholm, Sweden, 2011

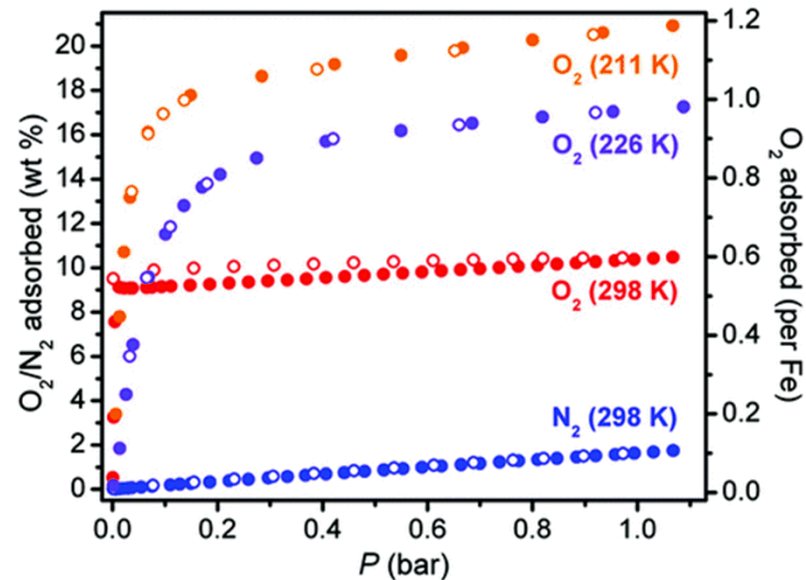
Literature MOFs Separate O₂/N₂



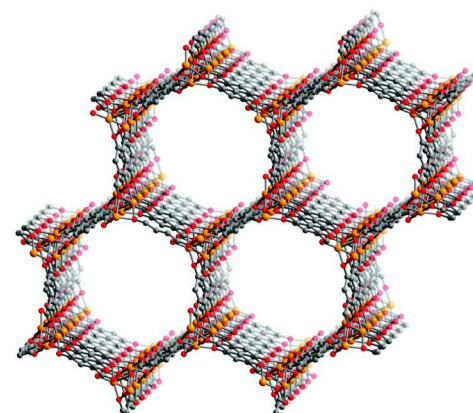
Cr₃(btc)₂



Long, J. R., et.al., *J. Am. Chem. Soc.* **2010**, 132, 7856.



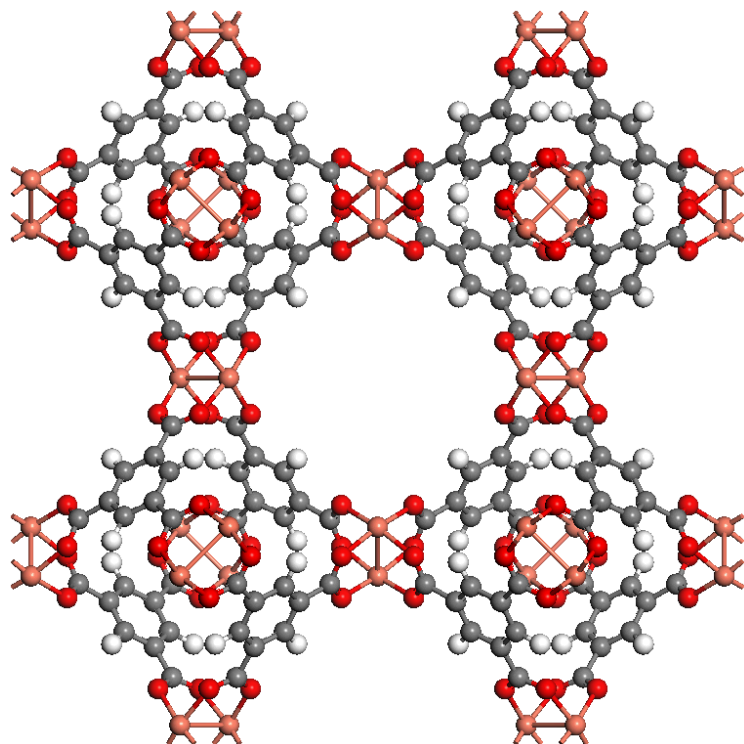
Fe₂(dobdc)



Long, J. R. et.al, *J. Am. Chem. Soc.* **2011**, 133, 14814

MOFs for O₂/N₂ Separation

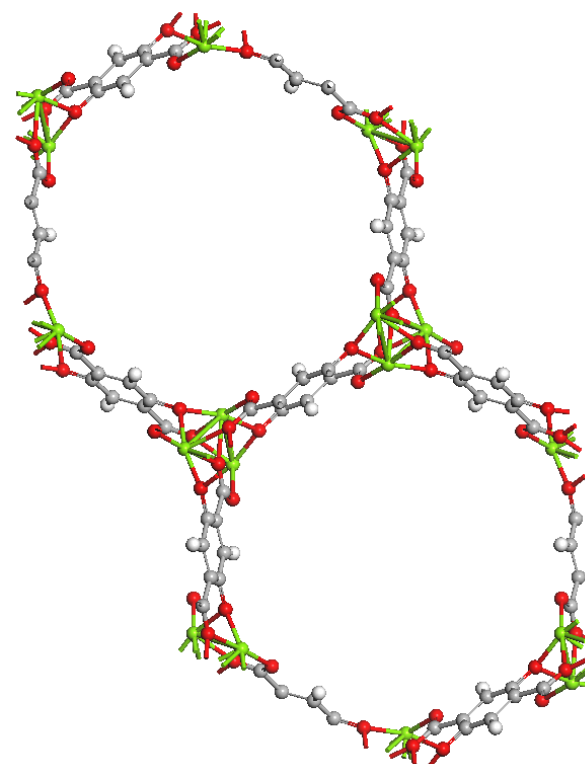
Cr₃(btc)₂



Cages: 6.9 Å, 11.1 Å, 13.2 Å

Windows: 4.1 Å, 6.9 Å

Fe₂(dobdc)



Channels: 10.8 Å

21	22	23	24	25	26	27	28	29	30
Sc	Ti	V	Cr	Mn	Fe	Co	Ni	Cu	Zn

Vienna *ab initio* simulation package (VASP)

- MOF
 - $M_3(\text{btc})_2$ analogs
 - $M_2(\text{dobdc})$ analogs
 - Metals
- | | | | | | | | | | |
|----|----|----|----|----|----|----|----|----|----|
| 21 | 22 | 23 | 24 | 25 | 26 | 27 | 28 | 29 | 30 |
| Sc | Ti | V | Cr | Mn | Fe | Co | Ni | Cu | Zn |
- Gas (one molecule per primitive cell)
 - O_2
 - N_2
 - Plane wave DFT geometry optimizations, periodic structure
 - PBE density functional with dispersion correction (PBE-D2), PAW potentials for core electrons
 - Spin polarization

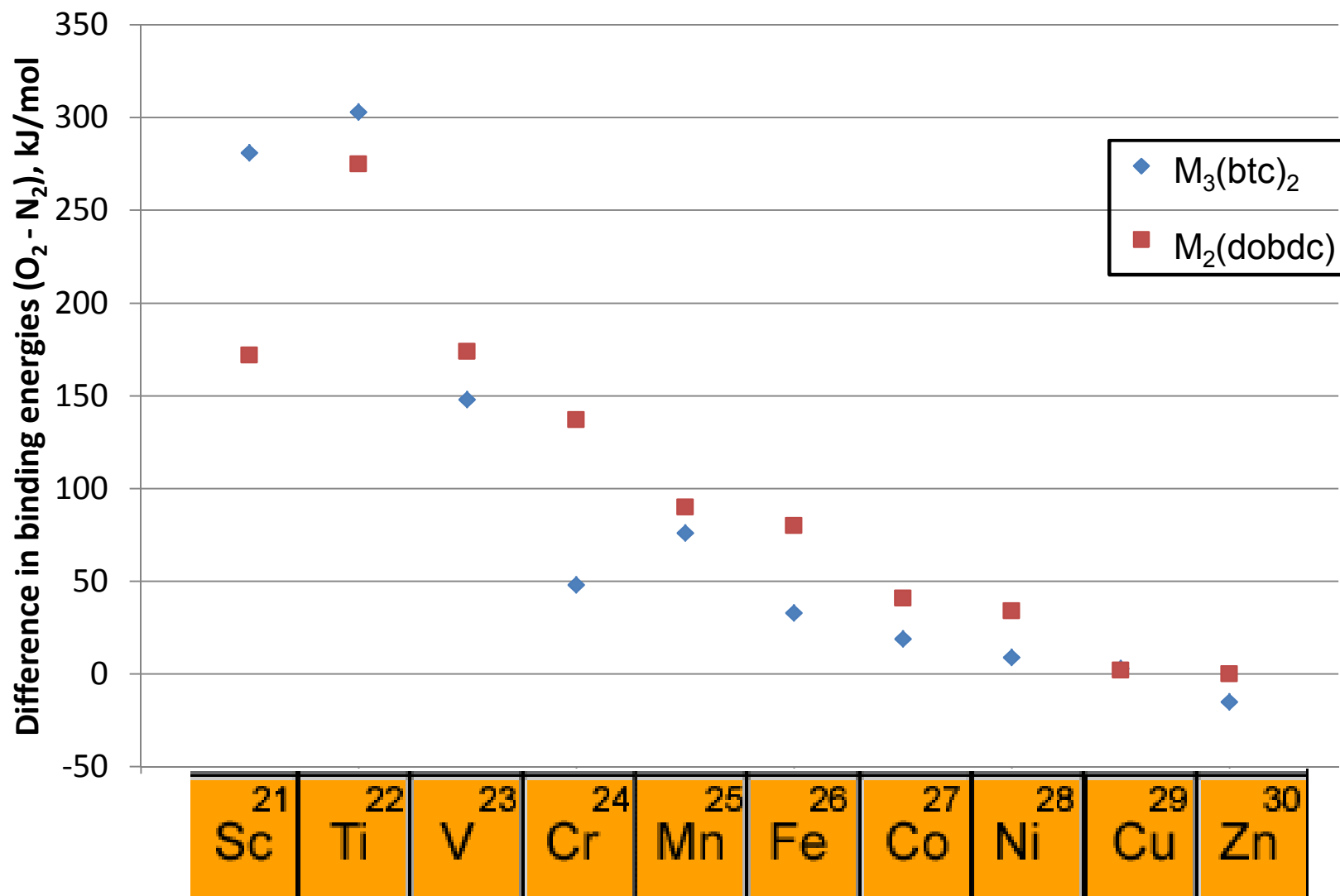
$$E_{\text{binding}} = E_{\text{MOF} + O_2} - (E_{\text{MOF}} + E_{O_2})$$

O₂, N₂ Binding Energies (ΔE), $\Delta \Delta E = \Delta E_{N_2} - \Delta E_{O_2}$

metal	M ₃ (btc) ₂ binding energy (kJ/mol)			M ₂ (dobdc) binding energy (kJ/mol)		
	O ₂	N ₂	$\Delta \Delta E$	O ₂	N ₂	$\Delta \Delta E$
Sc	-328	-46	281	-215	-43	172
Ti	-329	-26	303	-354	-79	275
V	-158	-10	148	-264	-90	174
Cr	-100	-52*	48	-147	-10	137
Mn	-264	-188	76	-116	-26	90
Fe	-45	-12	33	-88	-7	80
Co	-81	-62	19	-83	-41	41
Ni	-53	-44	9	-79	-46	34
Cu	-103	-100	3	-31	-29*	2
Zn	-22	-38	-15	-21	-21	0

* Gas atoms not adsorbed on metal

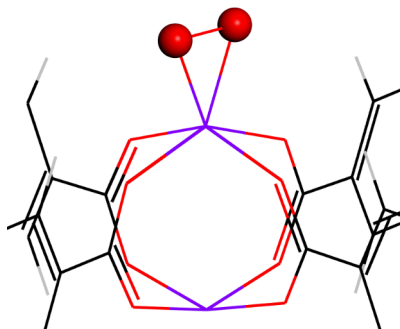
Difference in Binding Energies ($=\Delta\Delta E$)



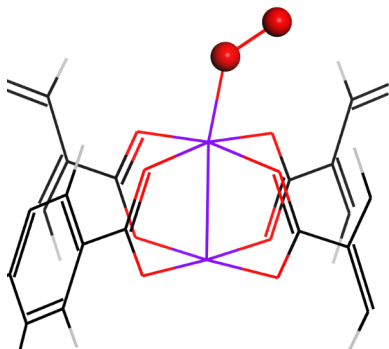
O₂ and N₂ Binding Modes

O₂

- Side-on

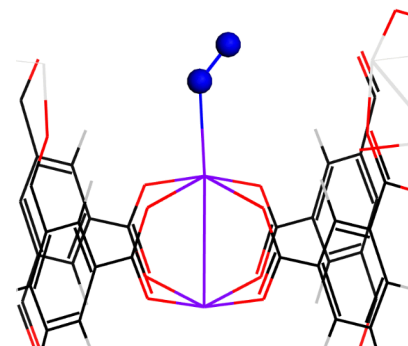


- Bent

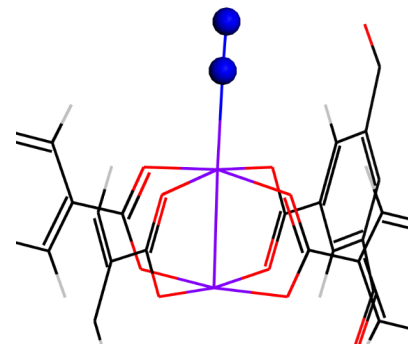


N₂

- Bent



- Linear



O₂, N₂ Binding Energies (ΔE): Highlighting M-MOFs easily made by metal center ion exchange

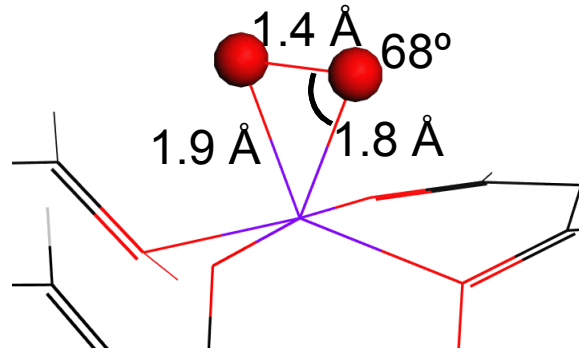
metal	M ₃ (btc) ₂ binding energy (kJ/mol)			M ₂ (dobdc) binding energy (kJ/mol)		
	O ₂	N ₂	$\Delta\Delta E$	O ₂	N ₂	$\Delta\Delta E$
Sc	-328	-46	281	-215	-43	172
Ti	-329	-26	303	-354	-79	275
V	-158	-10	148	-264	-90	174
Cr	-100	-52*	48	-147	-10	137
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Co	-81	-62	19	-83	-41	41
Ni	-53	-44	9	-79	-46	34
Cu	-103	-100	3	-31	-29*	2
Zn	-22	-38	-15	-21	-21	0

* Gas atoms not adsorbed on metal 11

J. Phys. Chem. C **2015**, 119, 6556.

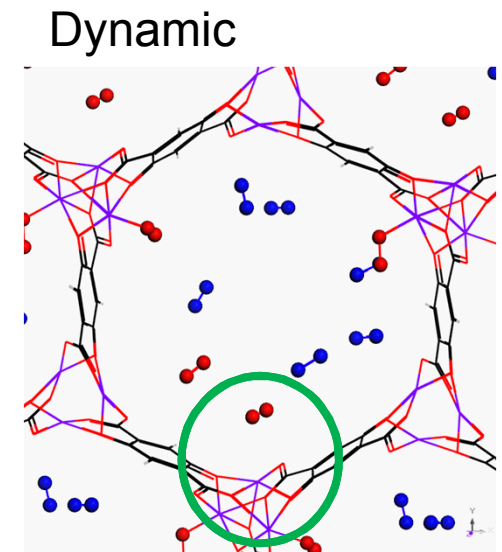
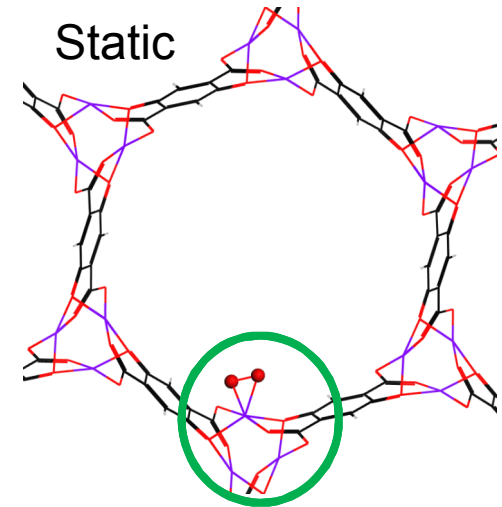
Minimum Energy Binding Mode

O₂ binding to Cr₂(dobdc)



- Side-on bonded, but not showing thermal bent geometry at dynamic binding
- -147 kJ/mol
- *Static configuration*
- 0 K

Minimum-energy configuration doesn't tell the whole story! "Snap Shot", no thermal effect



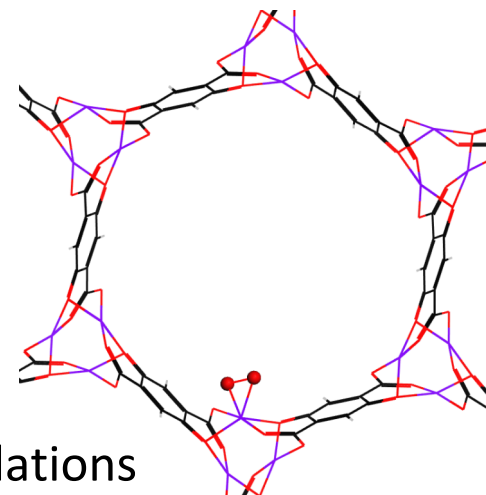
AIMD Calculations

Vienna *ab initio* simulation package (VASP)

- MOF
 - $M_2(\text{dobdc})$ analogs
- Metals

21	22	23	24	25	26	27	28	29	30
Sc	Ti	V	Cr	Mn	Fe	Co	Ni	Cu	Zn

- Temperatures
 - 201 K
 - 258 K
 - 298 K



- AIMD simulations
 - NVT ensemble
 - 27.5 ps
 - 0.5 fs timestep
- PBE density functional with dispersion correction (PBE-D2), PAW potentials for core electrons, spin polarization

Parkes, M.V.; Sava Gallis, D. F.; Greathouse, J.A.; Nenoff, T.M. *PCCP*, **2016**, 18, 11528.

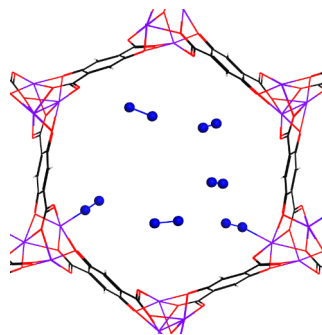
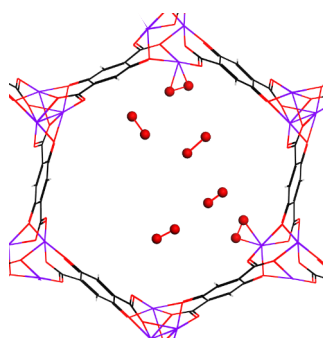
Initial Gas Positions From Static DFT

Guests

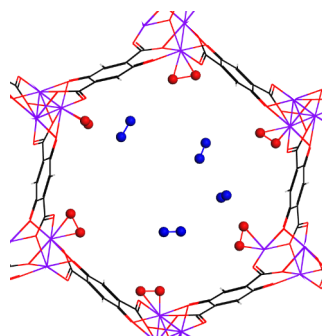
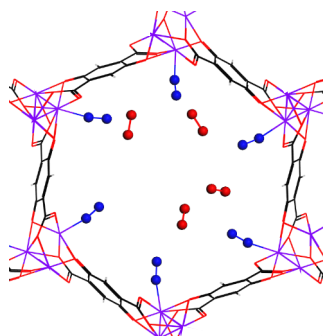
2 O₂ bound
4 O₂ unbound

2 N₂ bound
4 N₂ unbound

Single
component



Mixed gas
**Competitive
binding**



6 N₂ bound
4 O₂ unbound

6 O₂ bound
4 N₂ unbound

Temperatures

201 K
258 K
298 K

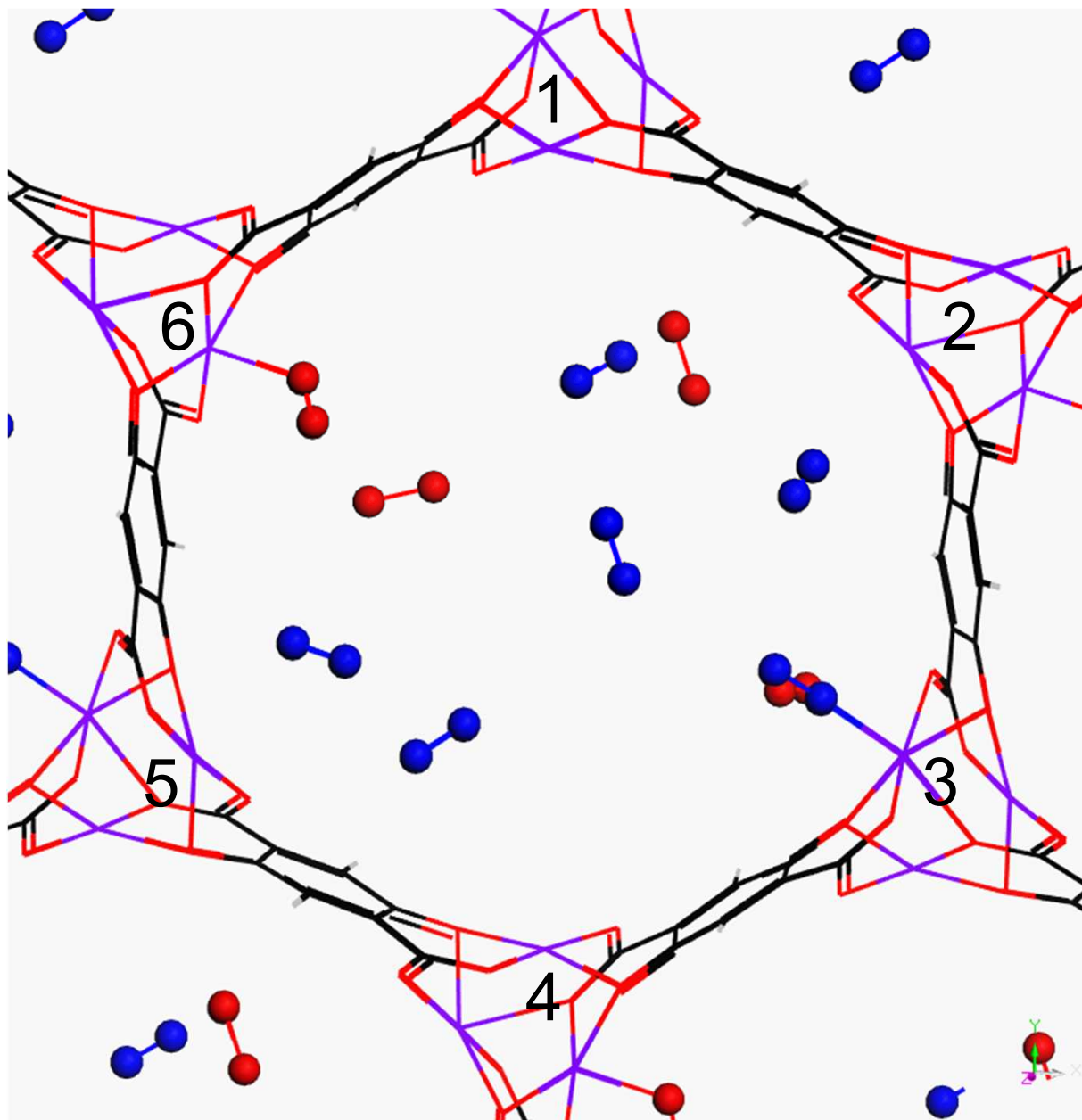
Metals

24	25	26
Cr	Mn	Fe

Red Sky Supercomputer
36 Simulations

3,800 processor-days each

<http://hpc.sandia.gov/>



Cr₂(dobdc)
6 N₂ + 4 O₂
298 K

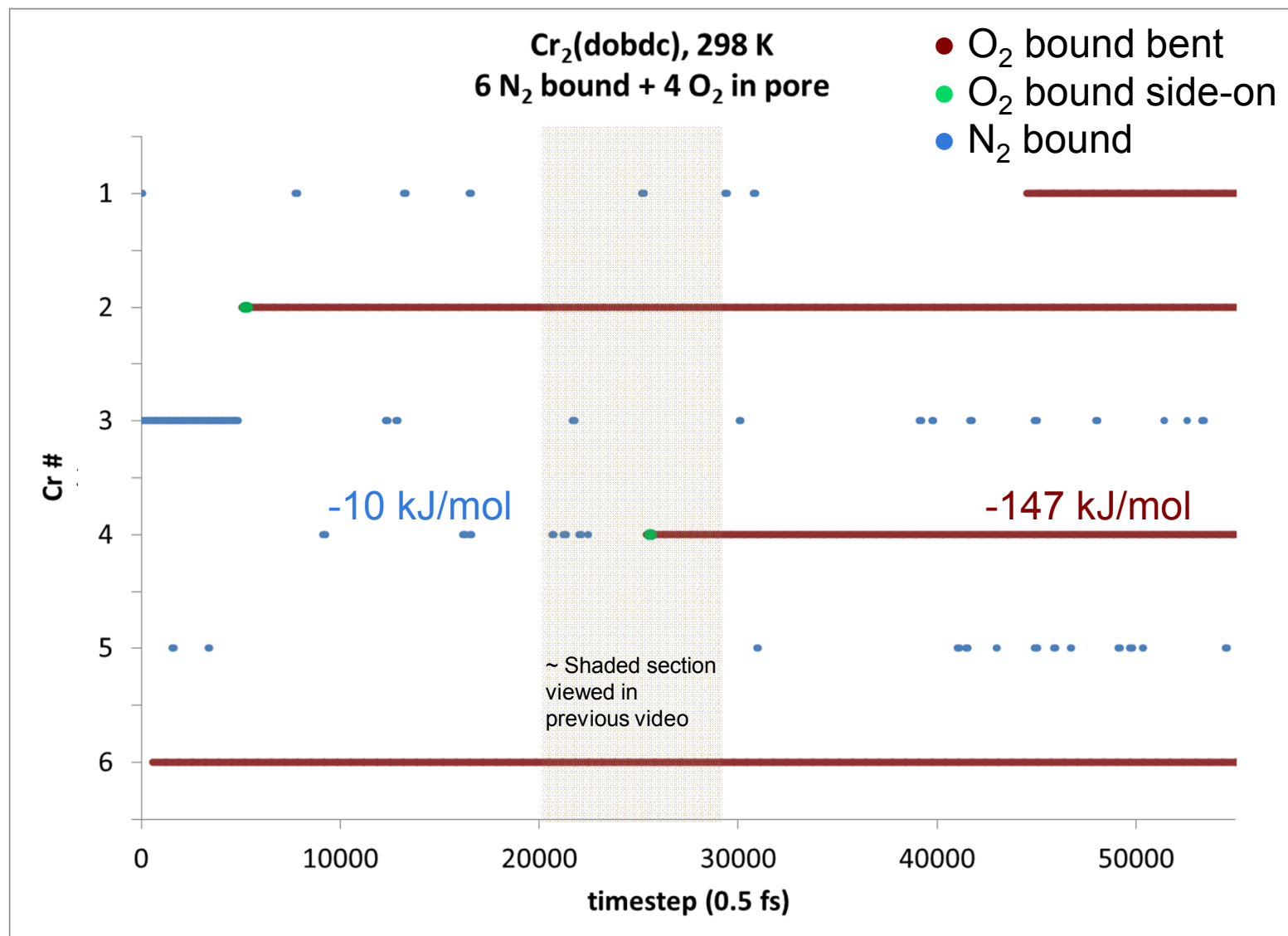
NVT
Time 2 ps – 15 ps

1 frame = 25 fs

- O₂ slow to bind, but once on metal center, binding holds
- N₂ rapid bind and release from metal centers
- O₂ long term binding is consistently 'bent'
- Selective for O₂

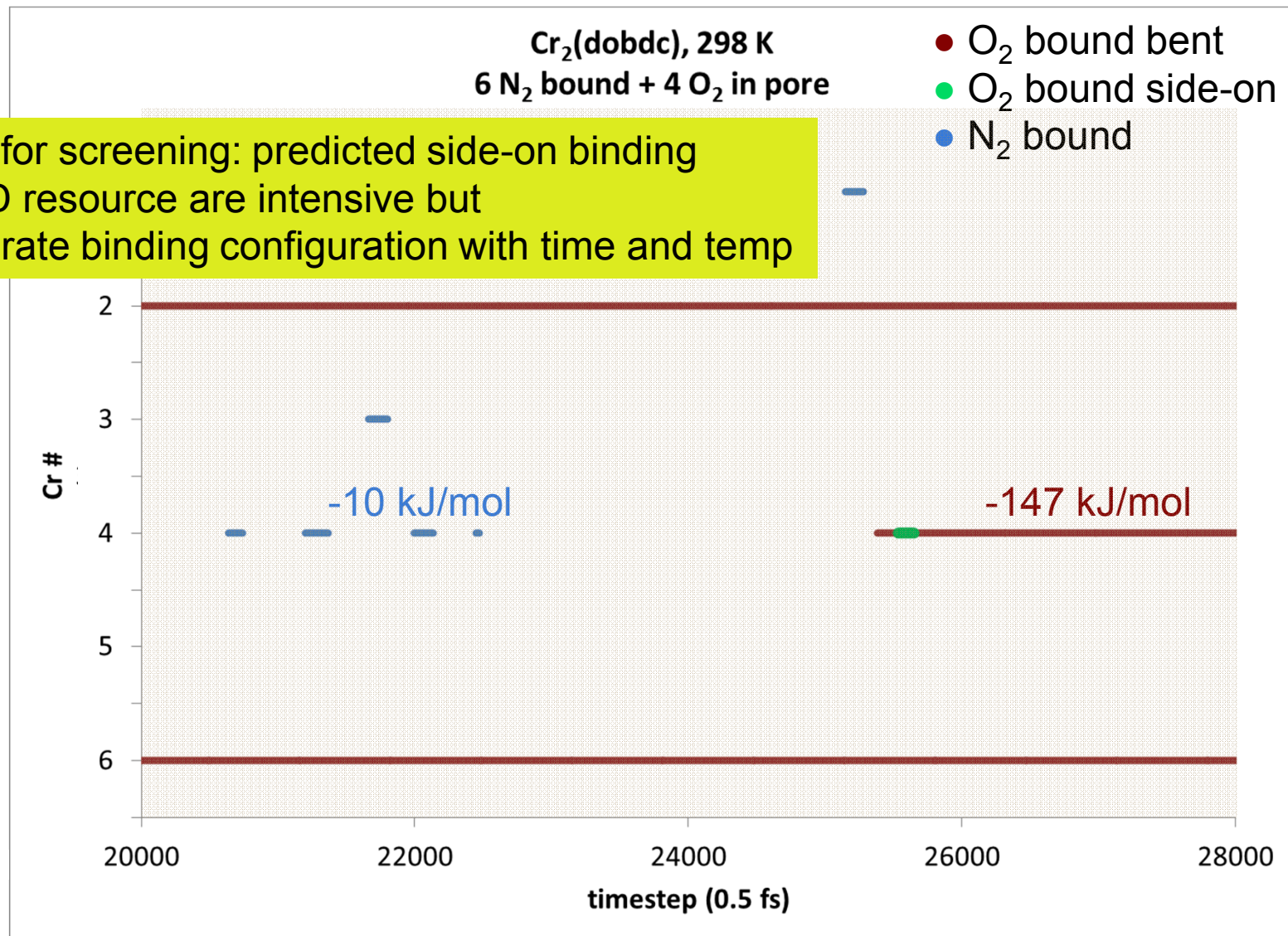
PCCP, **2016**, 18, 11528

Gas Occupancy at Each Metal Site



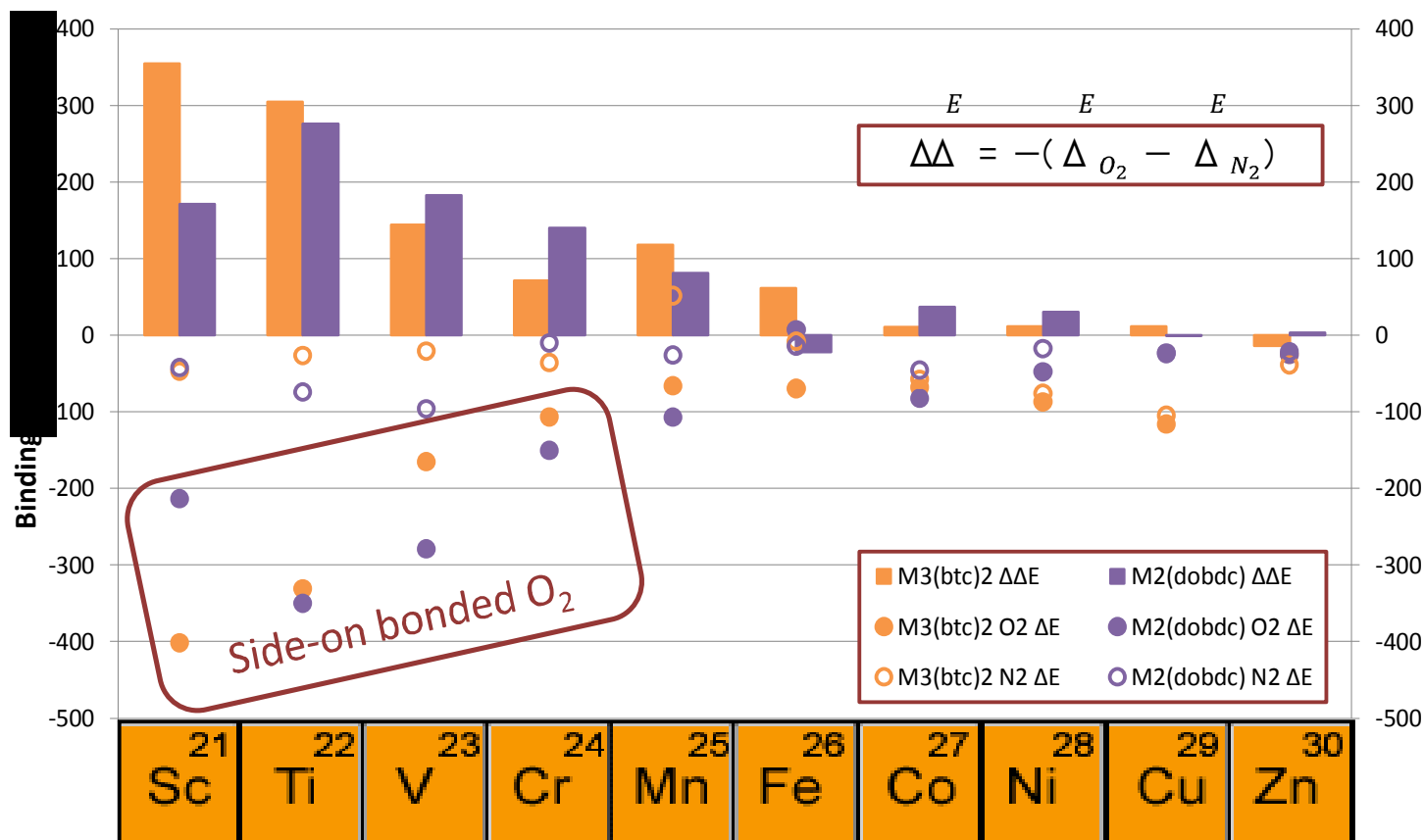
Gas Occupancy at Each Metal Site: *previous slide shaded area*

DFT for screening: predicted side-on binding
AIMD resource are intensive but
Accurate binding configuration with time and temp



Study of Early Transition Metal MOFs for O₂ Selectivity

Binding Energy Calculated as a Function of Metal Site



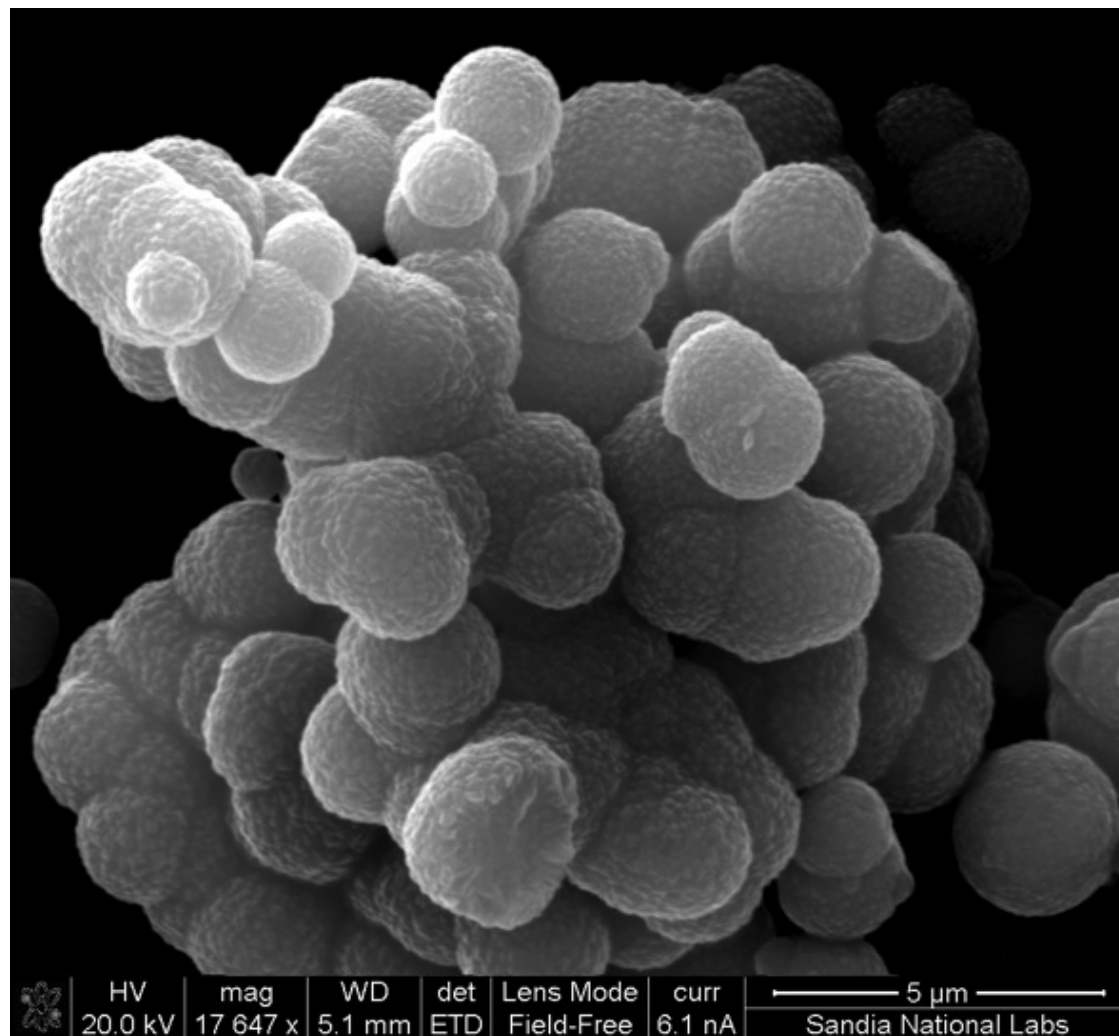
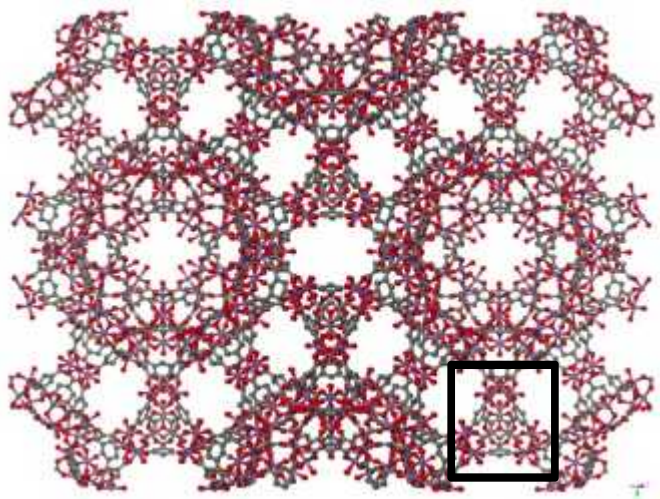
Sc-MIL-100 for O₂ Selectivity

Sc/BTC/DMF/HCl

Unique synthesis:

Mixed Sc(NO₃)₃•xH₂O and
1,3,5-benzetricarboxylic acid
in N,N'-dimethylformamide and
HCl.

Heated to 373K overnight



Sava Gallis, D. F., Greathouse, J.A.; Rodriguez, M.A.; Parkes, M.V.; Chapman, K.W.;
Nenoff, T.M. *Chem. Mater.* **2016**, 28(10), 3327.

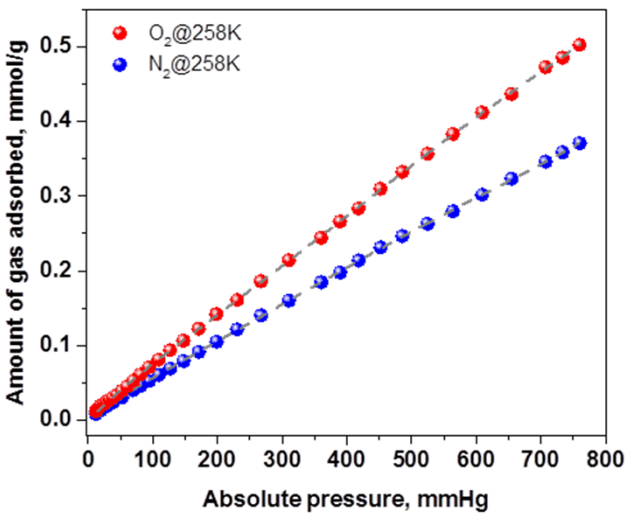
US Provisional Patent 2015

19

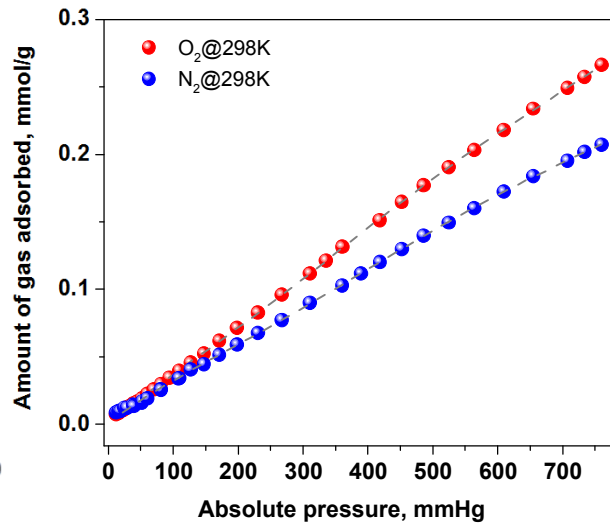
Sc-MIL-100: Enhanced Quantity of O₂ vs N₂ Adsorbed over Wide Temperature Range (at least to 313K)

— Fit using the virial eq.

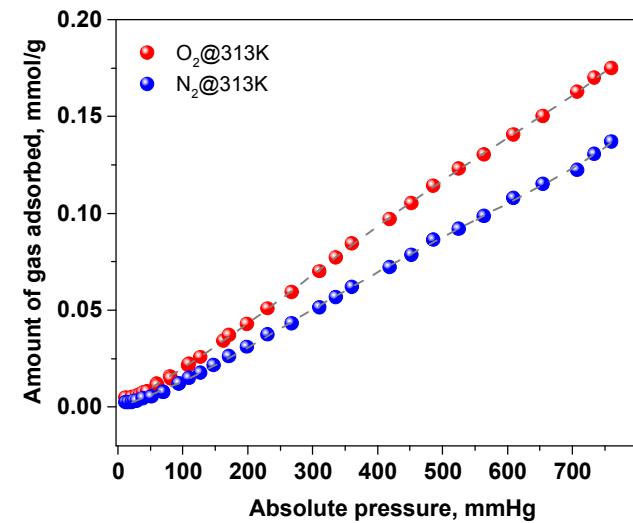
O₂ vs. N₂ @258K



O₂ vs. N₂ @298K



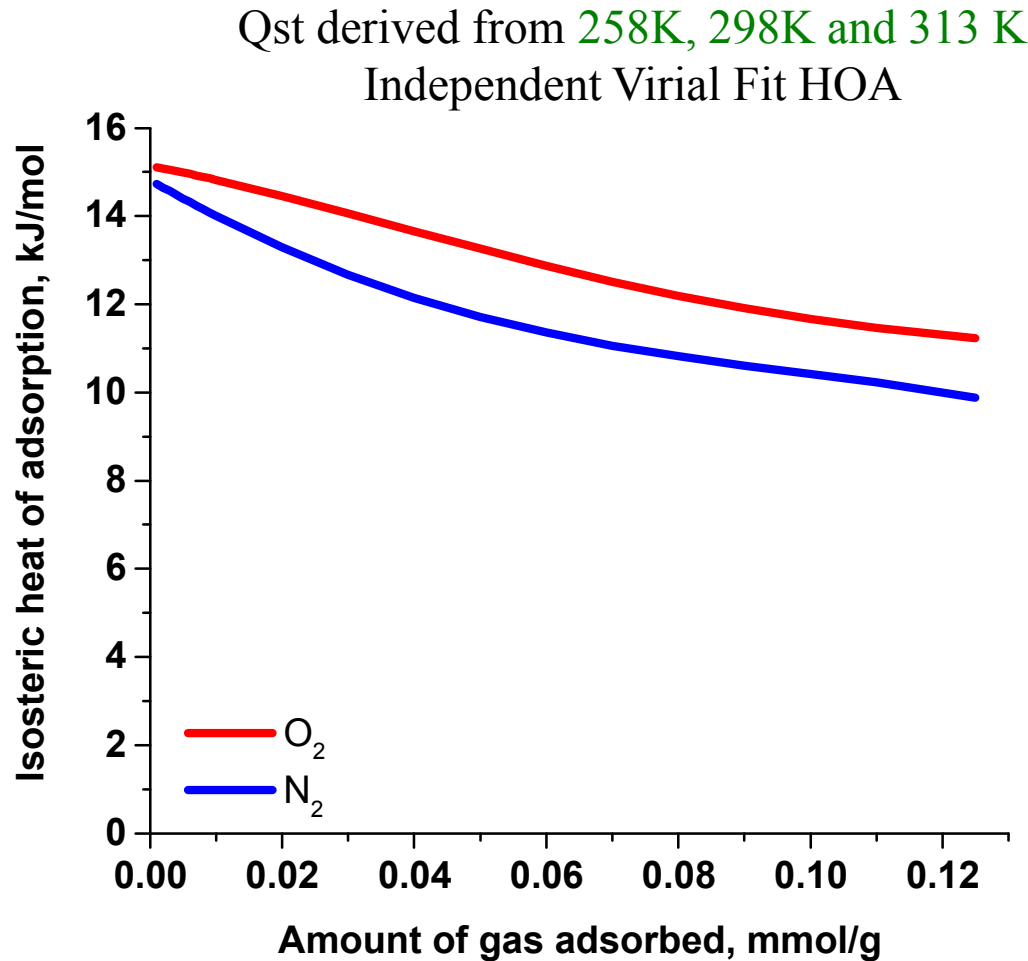
O₂ vs. N₂ @313K



Isotherm trends mimic those predicted by GCMC

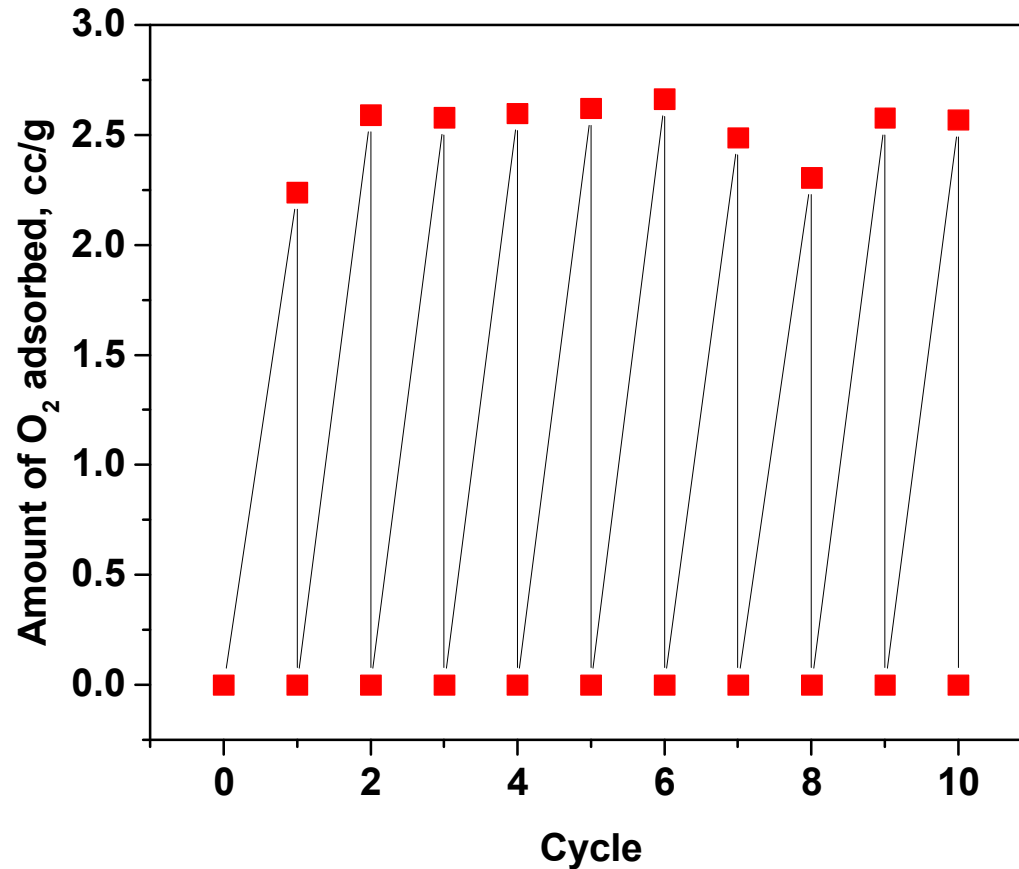
Isosteric heat of Adsorption (kJ/mol)

Higher binding energy for O₂ vs N₂

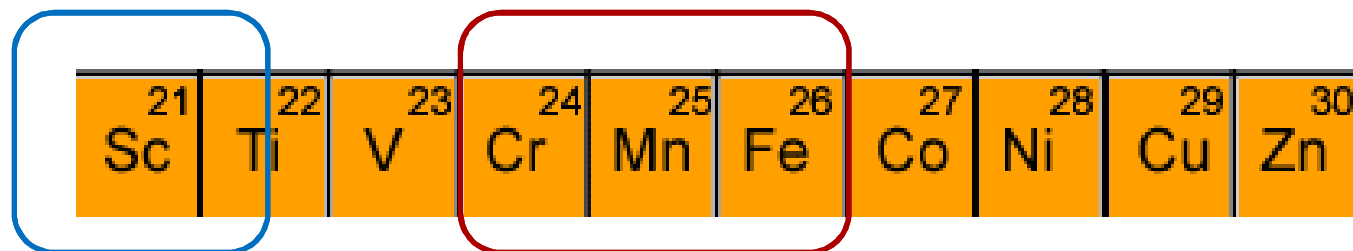


US provisional Patent 2015

O₂ Adsorption and Desorption over 10 cycles, 298K, 1 atm



Conclusions



21	22	23	24	25	26	27	28	29	30
Sc	Ti	V	Cr	Mn	Fe	Co	Ni	Cu	Zn

- Metal matters!
 - Mid transition metal trends are consistent in two different MOF frameworks
- Focus on mid transition metals
 - Easy to synthesize
 - Mild yet preferential O₂ binding
- Novel use of AIMD
 - Use AIMD to obtain more insights into mechanism of binding
- Differences in O₂/N₂ binding to metal sites
 - O₂ binds and sticks
 - N₂ binds on-off
- Early Transition Metals: Open MIL-100 framework, Sc shows optimum binding for O₂

Thank You

Sandia National Laboratories' Sites



Albuquerque, New Mexico



**Kauai Test Facility
Hawaii**



**Tonopah Test
Range,
Nevada**



**Yucca Mountain,
Nevada**



**WIPP,
New Mexico**



Pantex, Texas



Livermore, California