

Applications of EBSD in Materials Science

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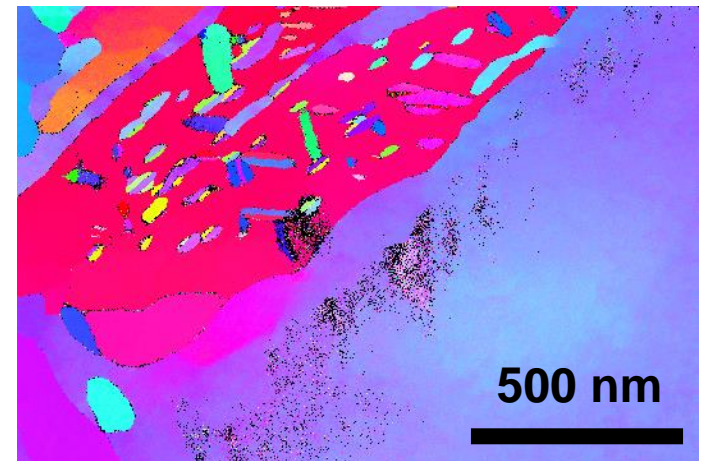
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Gibeon Meteorite – Microstructure over a huge difference in length scales



EBSD and TKD allow a large range of length scales to be studied! We can collect data that covers mm and with nearly the same technique we can collect data on the nm scale!



Using EBSD “cleaning” routines – ethical or not?

“EBSD seems to be one of the few techniques in electron microscopy where the extensive interpolation and modification of data through filtering of cleaning routines may be accepted without careful comment about the process.”

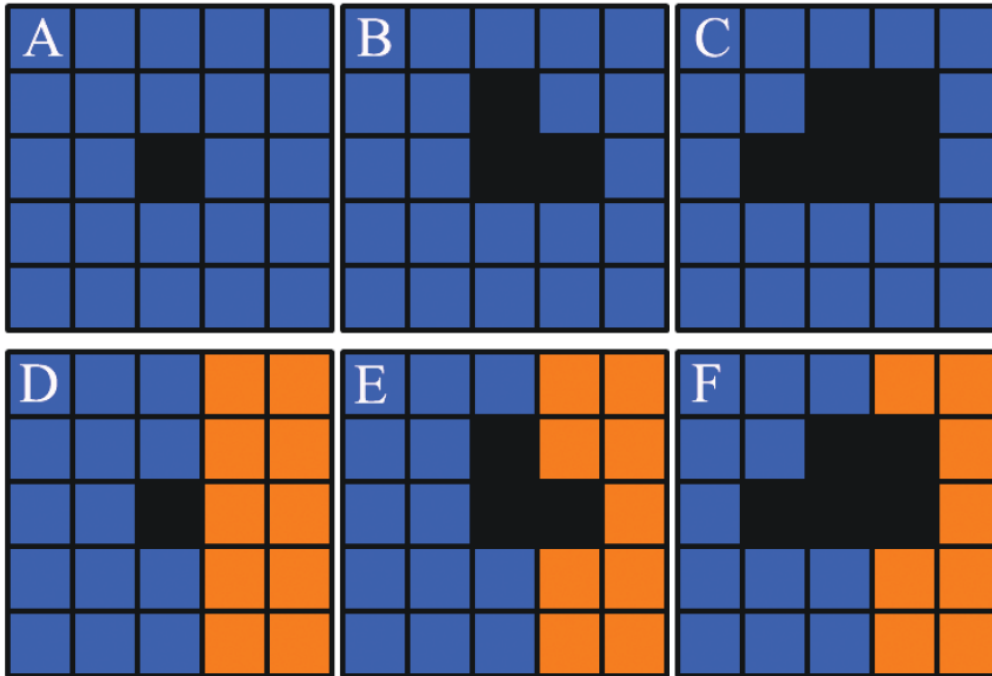
There are no perfect (at least I have not seen one) EBSD maps with no missing or mis-indexed pixels as acquired, but there are many in the literature.

How many of these papers with perfect EBSD maps tell you what they did to make them that way?

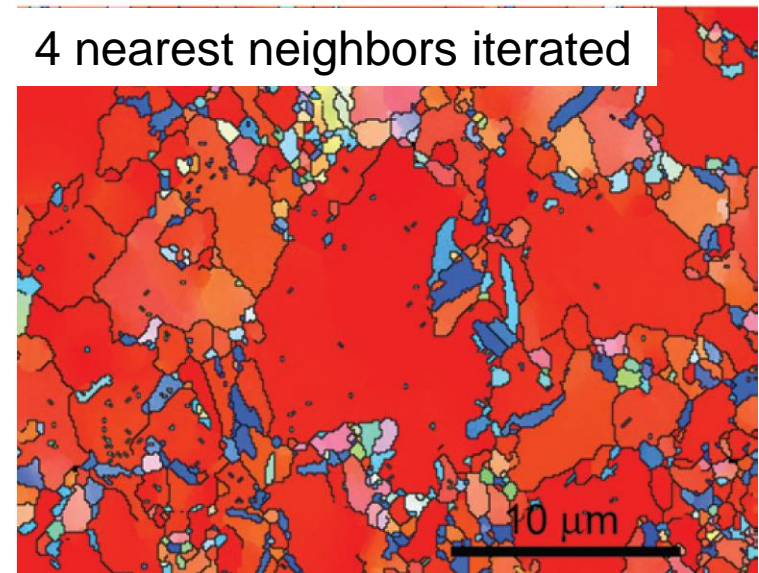
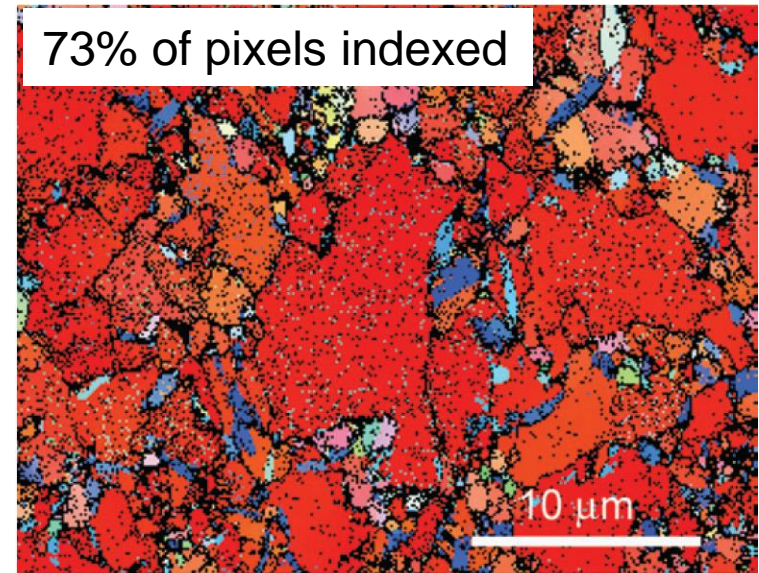
The vendors of EBSD tools encourage this by supplying the “Easy button” that removes random mis-indexed pixels and then fills in the pixels that do not get indexed.

***Brewer and Michael, Risks of “cleaning” EBSD data, Microscopy Today, 2010, March, p. 10-15.**

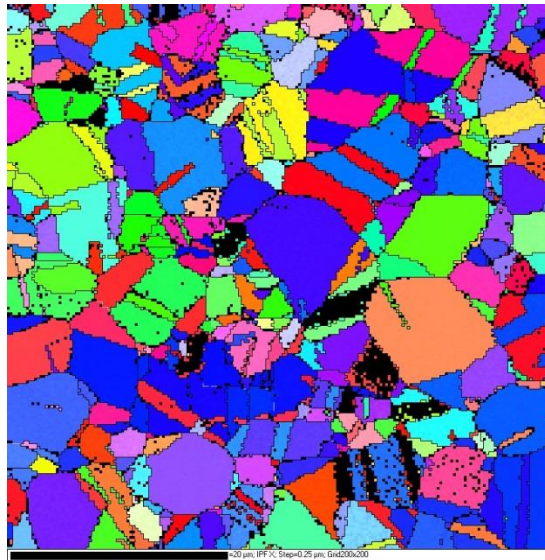
EBSD cleaning routines



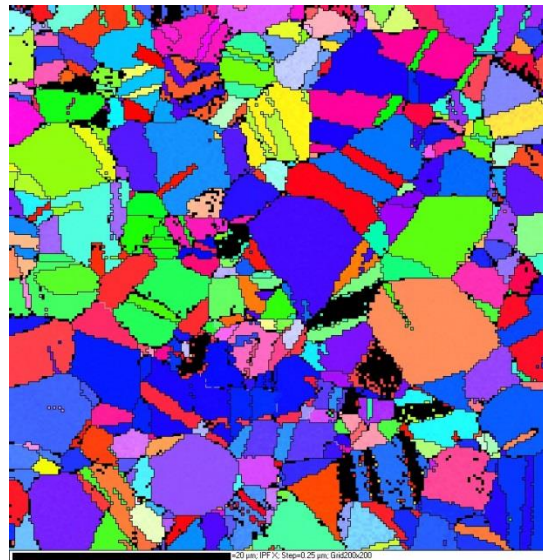
Be careful how the nearest neighbor hole filling routines are applied – can end up with some odd looking microstructures.



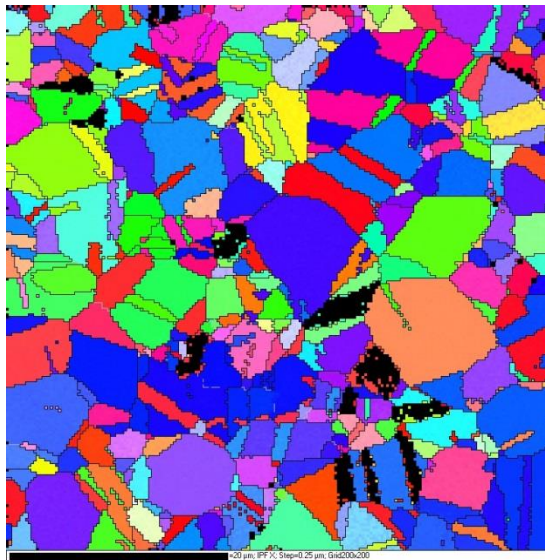
**Raw
data**



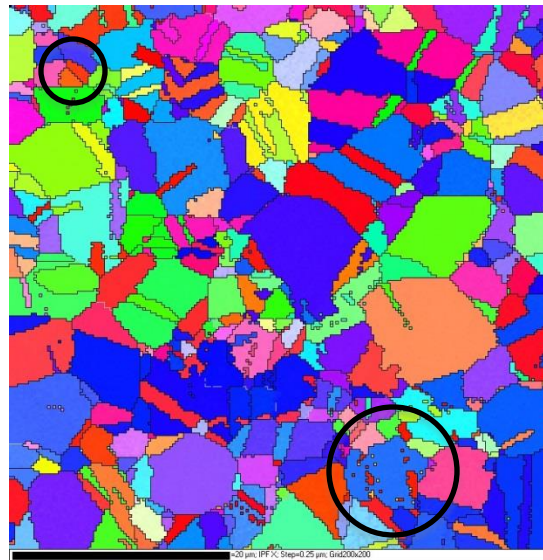
8 neighbors



6 neighbors

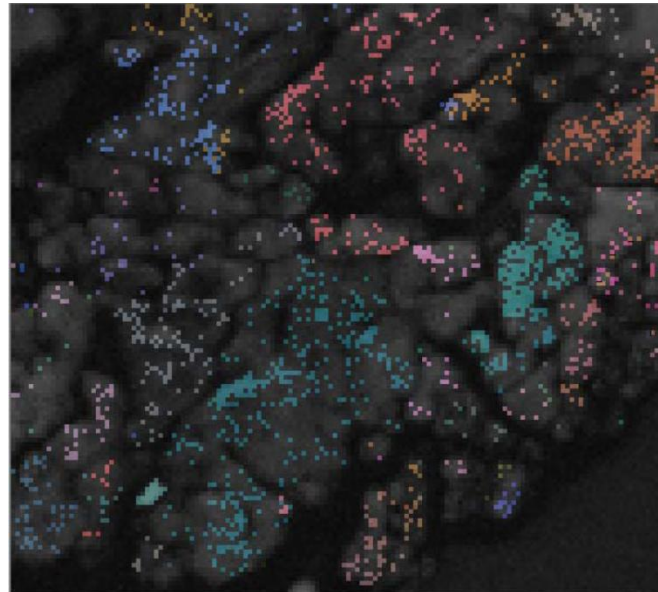
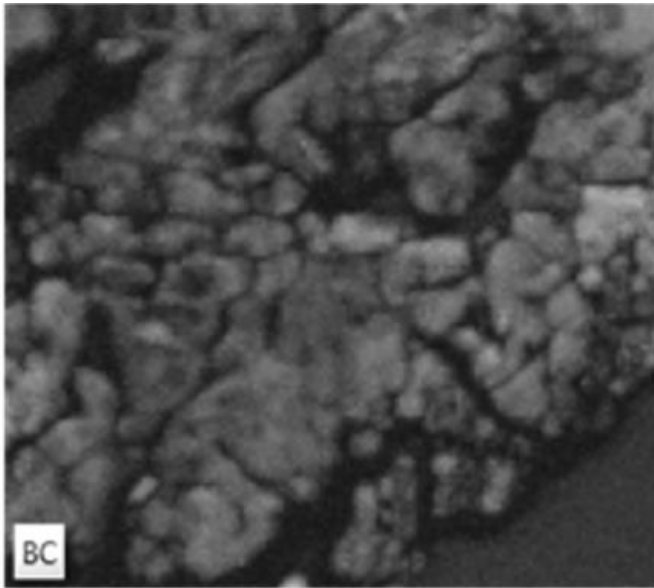


4 neighbors



Impact of kernel size used to fill in pixels.

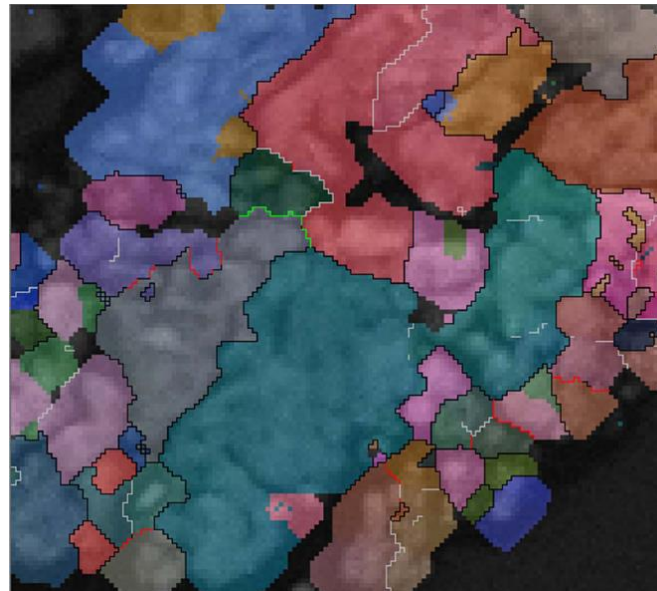
Extreme data cleaning -



Actual indexed
pixels

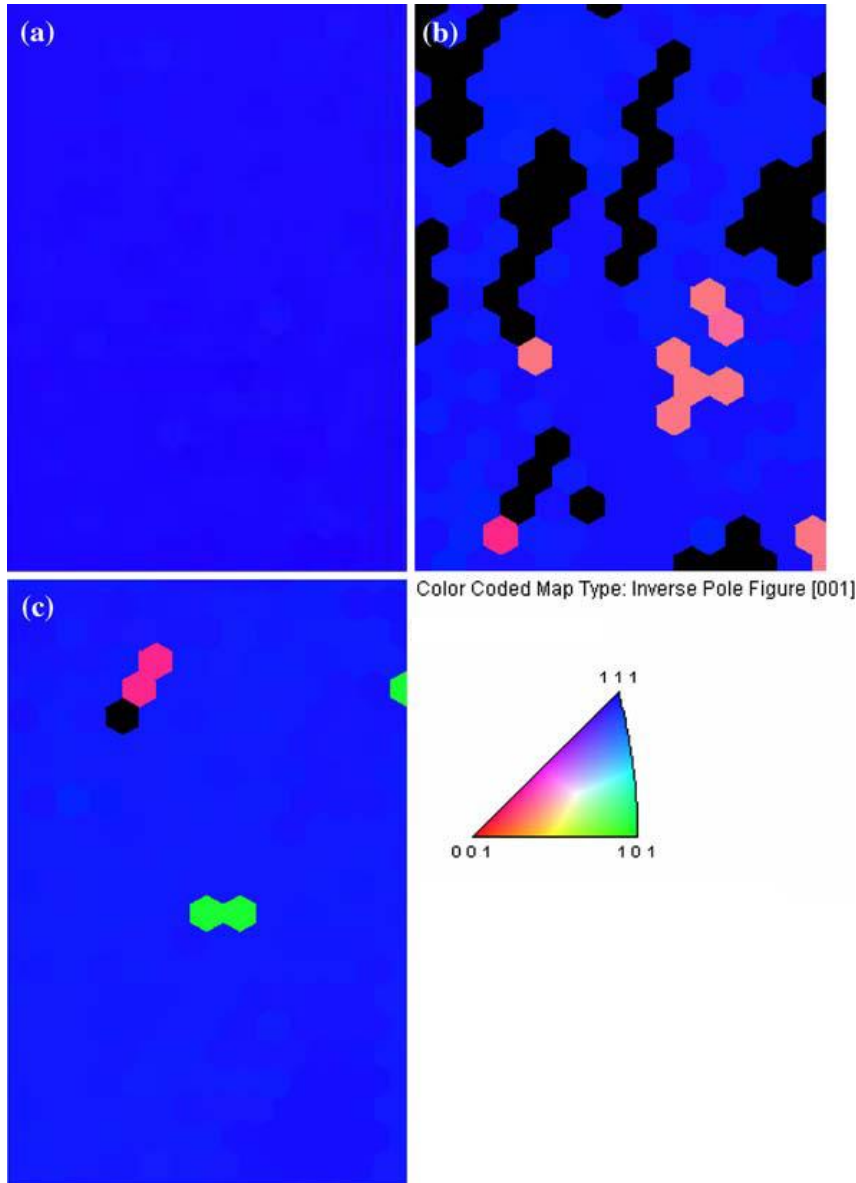
TKD example of what I
consider to be “probably”
too much data cleaning.

Maybe make a better
sample of figure out why
there are issues.



Cleaned data!

Things we (as EBSD experts and peer reviewers) can do to ensure that EBSD remains a strong, vibrant and trusted technique



Why do maps like this get published?

No description of processing applied, orientation of sample or even a scale bar in the text or figure caption!

We must and we need to do better than this.

Using EBSD “cleaning” routines – ethical or not?

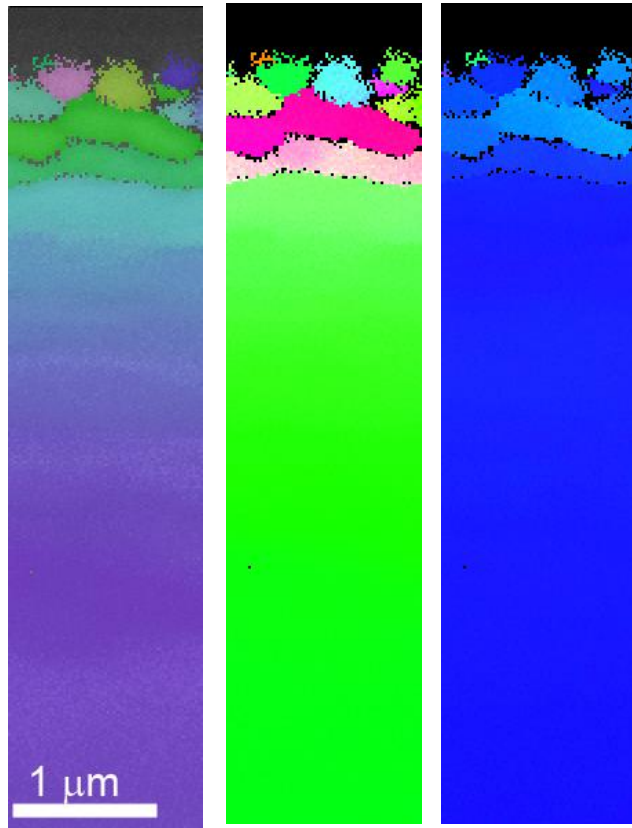
At the very least, each of us should fully disclose how the EBSD data were treated and why.

For example (and for many of the maps shown in this presentation):

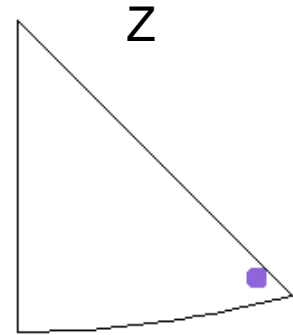
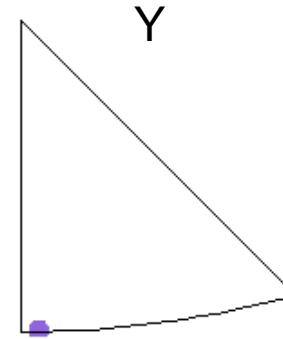
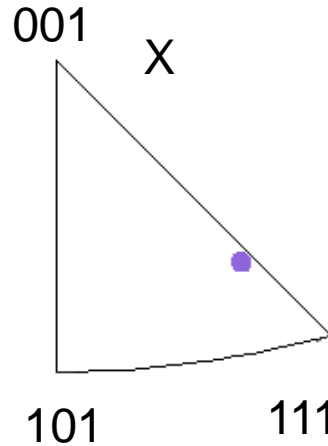
The EBSD data was:

- 1. Filtered to remove mis-indexed or rogue pixels**
- 2. An 8 nearest neighbors hole filling routine was used to fill in the pixels that were not indexed.**
- 3. If interfaces are important – more aggressive hole filling was applied to ensure that the grain boundaries or interphase interfaces were complete.**

Know your frames of reference!



Wear surface of Ni worn in the $\langle 211 \rangle$ on (101) plane

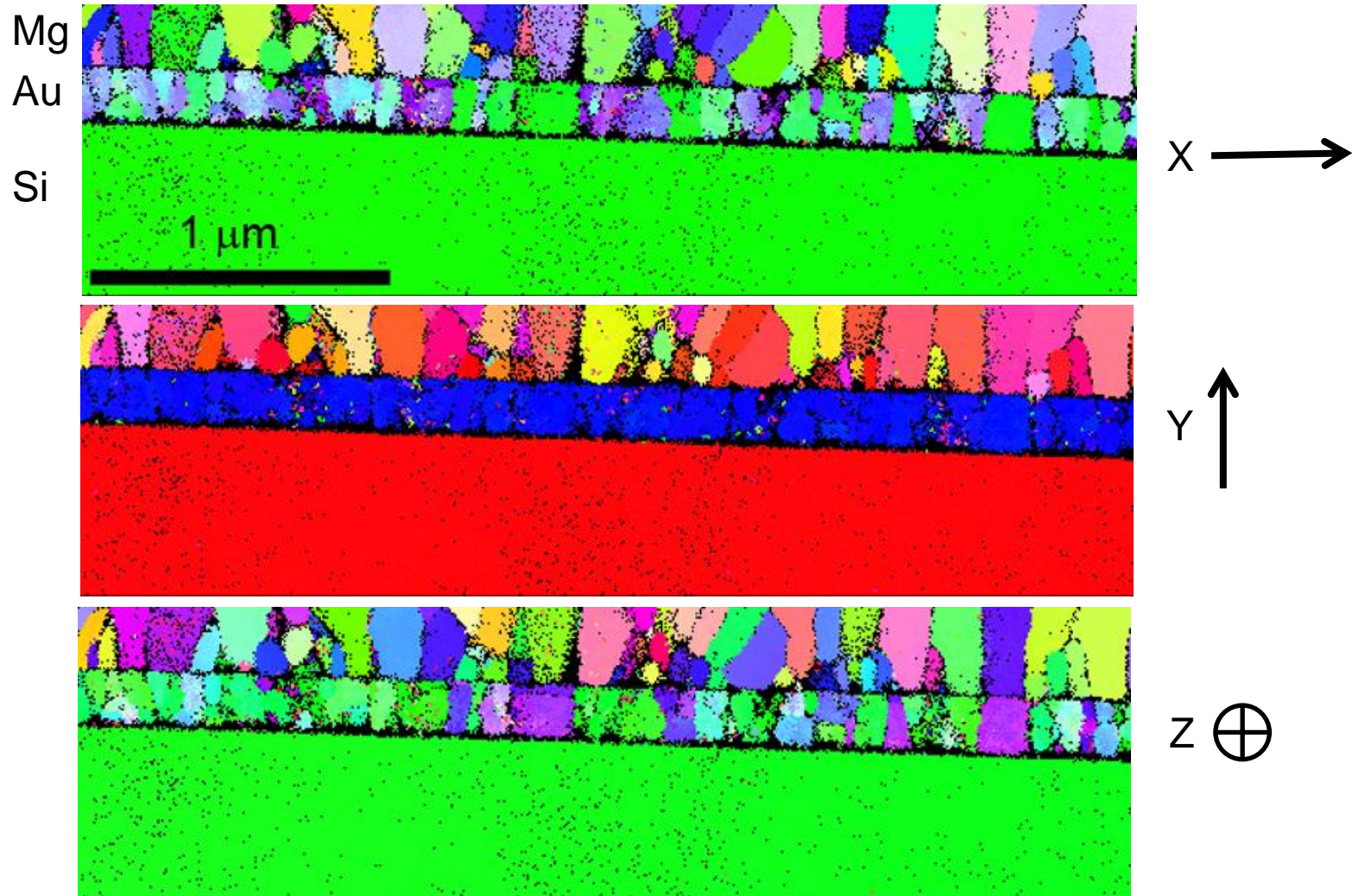


Periodically check to ensure that the data makes sense with respect to what we know about the sample.

In this case we knew we had a (101) single crystal surface and the wear scar was oriented along the $\langle 211 \rangle$. The orientation data supports this.

Know your frames of reference!

Au and Mg layers deposited on a (001) Si wafer

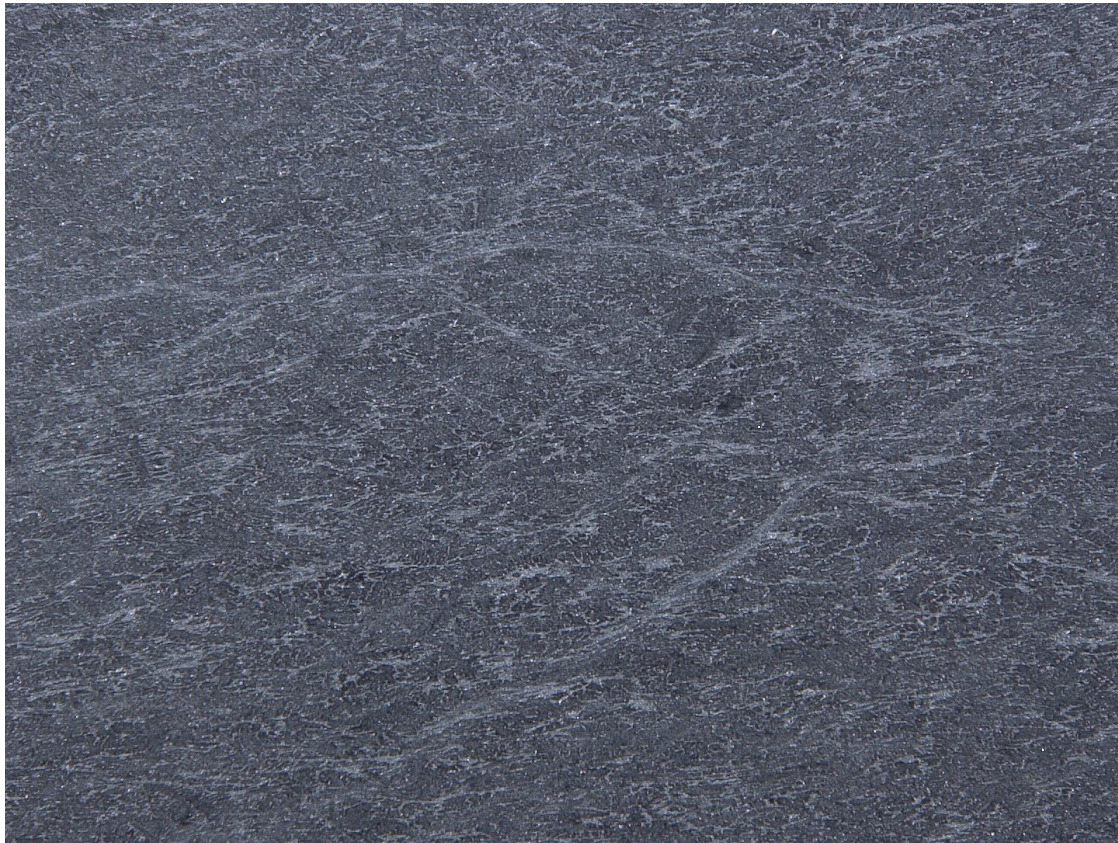


Wilkinson, A. J., T. B. Britton, J. Jiang, Y. Guo, A. Vilalta-Clemente, D. Wallis, L. N. Hansen, and A. Winkelmann. "Tutorial: crystal orientations and EBSD—or which way is up?." *Materials Characterization* (2016).

Analysis of linear feature in strained Al-Si

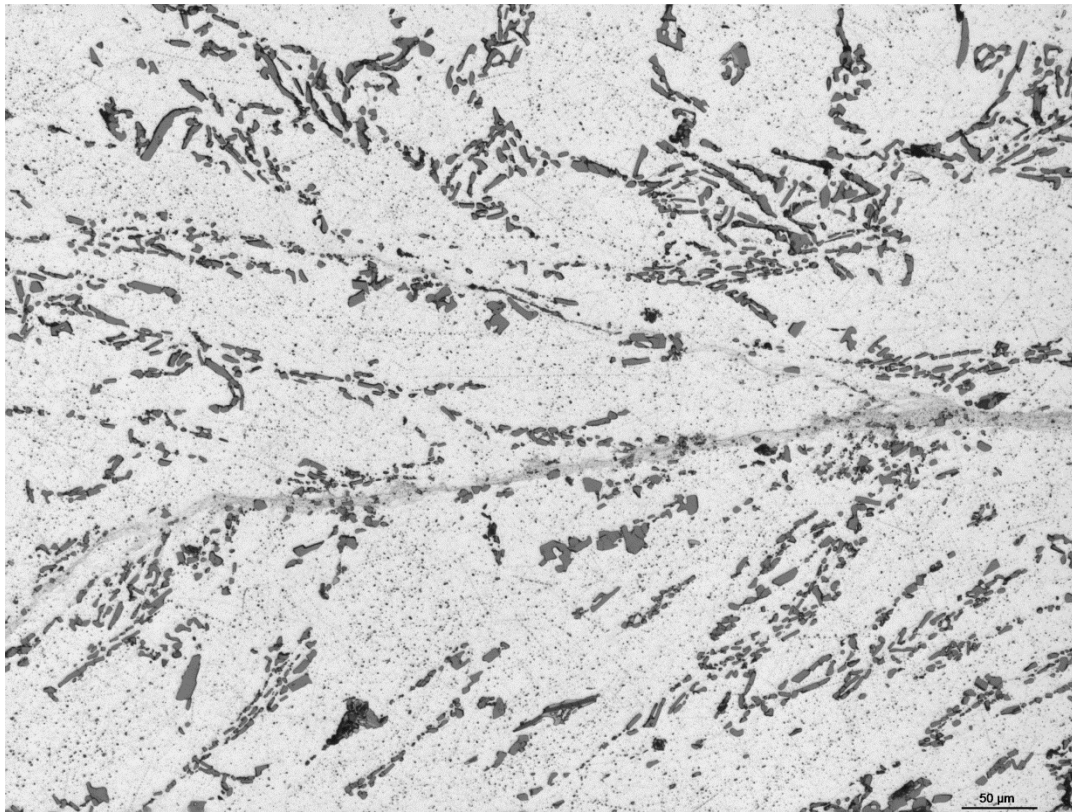
Sample was subjected to high strain rate deformation

Appearance of linear microstructural features – what are they?



Macrograph of etched surface of sample

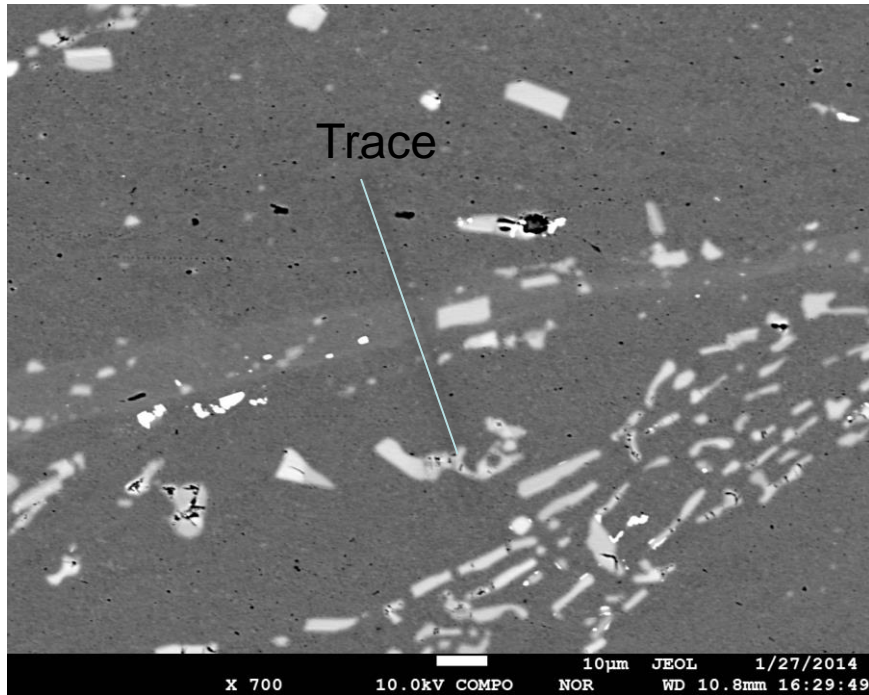
Adiabatic shear bands in an Al-Si alloy



What is this linear feature?

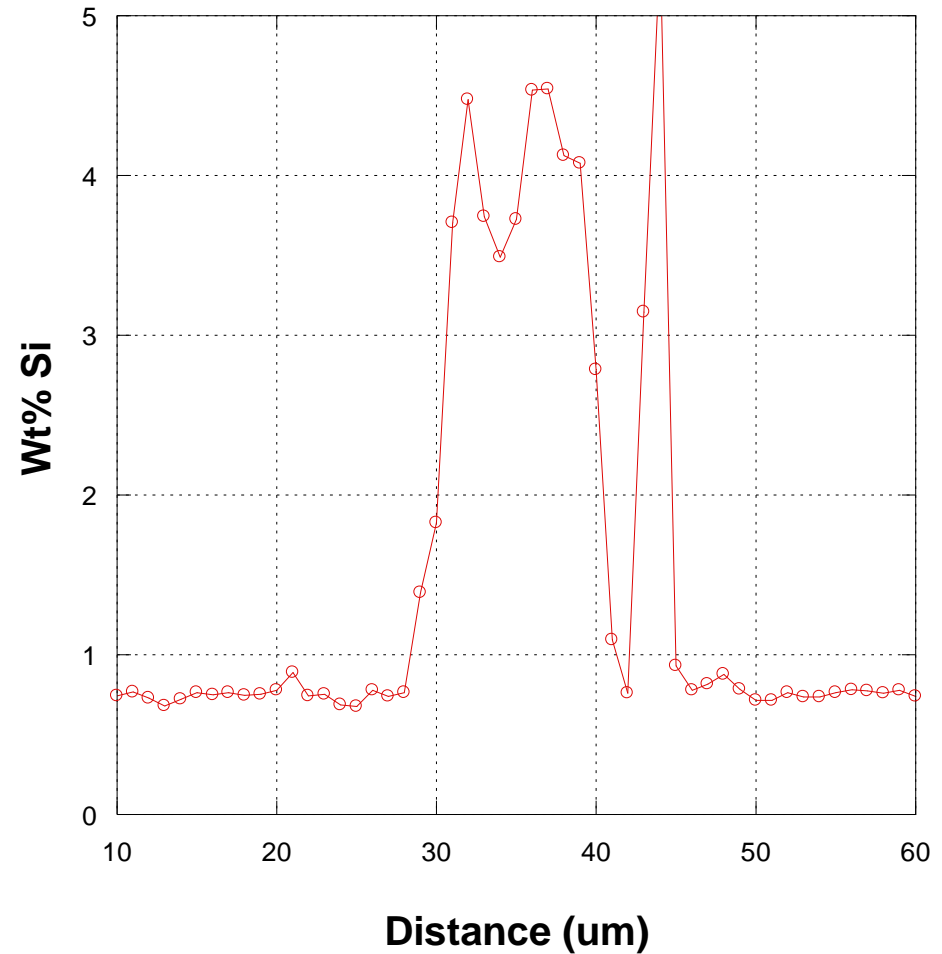
Adiabatic shear band – region of rapid high deformation where slip occurs and causes local temperature rise. Combination of heat and deformation results in dynamic recrystallization in ASB

Adiabatic shear bands in an Al-Si alloy



EPMA quantitative analysis shows an increase in Si in linear feature

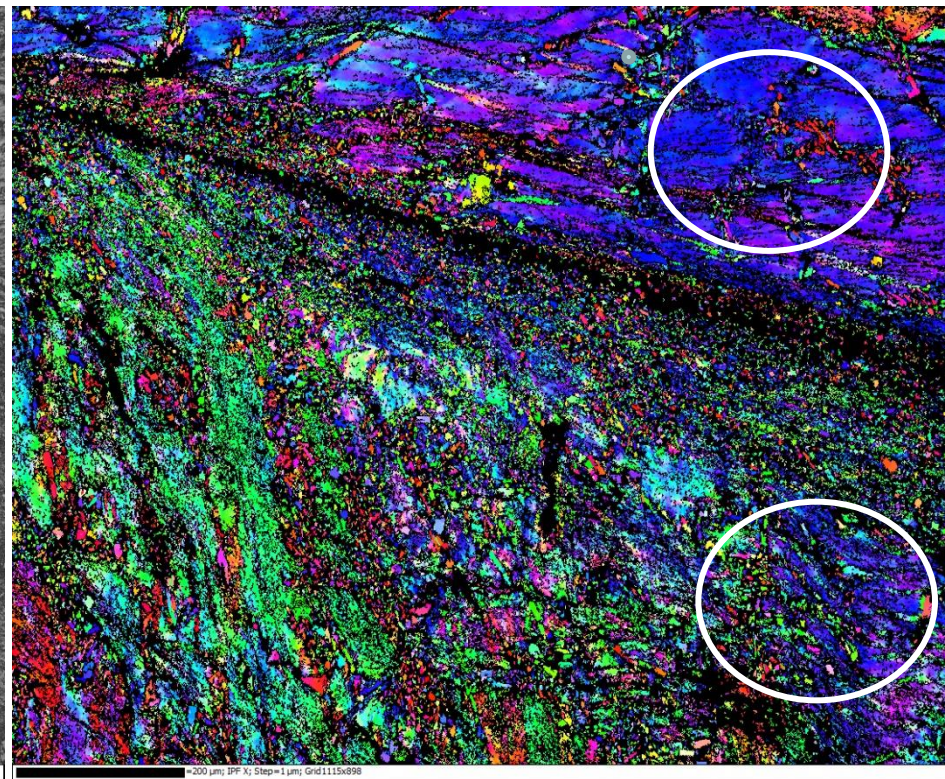
Consistent with local heating followed by rapid cooling.



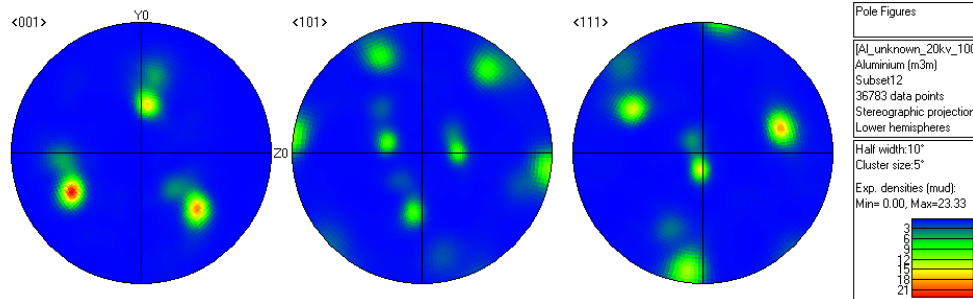
EBSD of deformed Al-Si alloy



Band Contrast

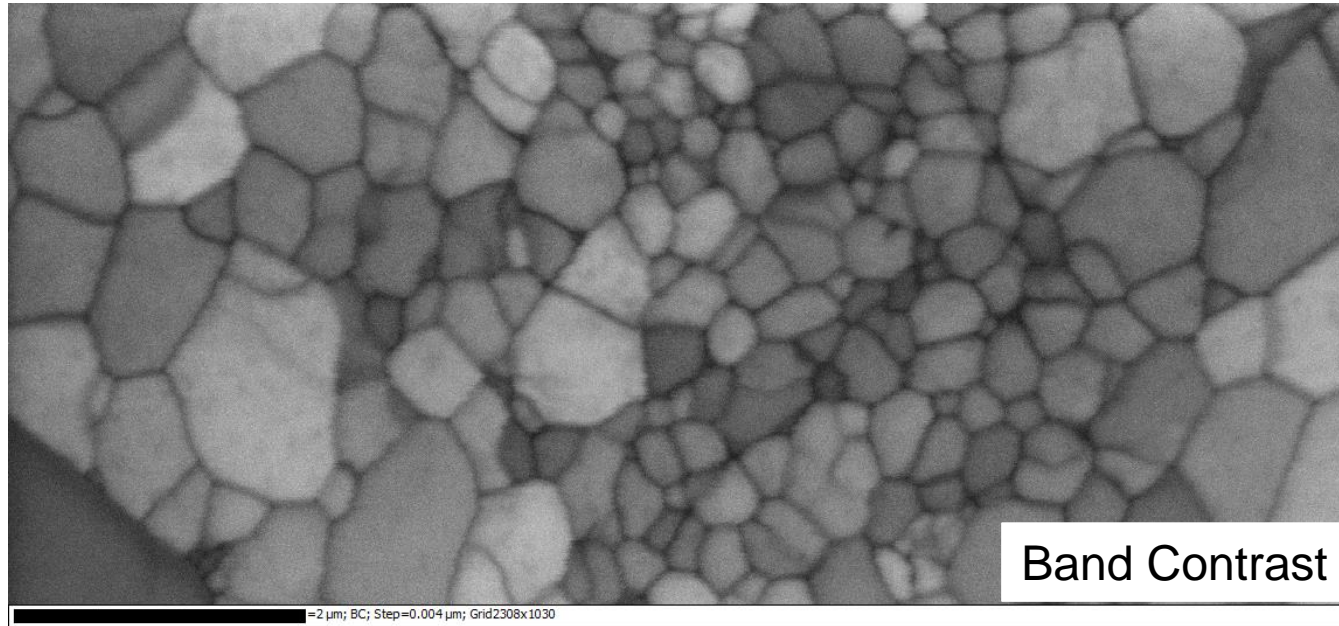


IPF X

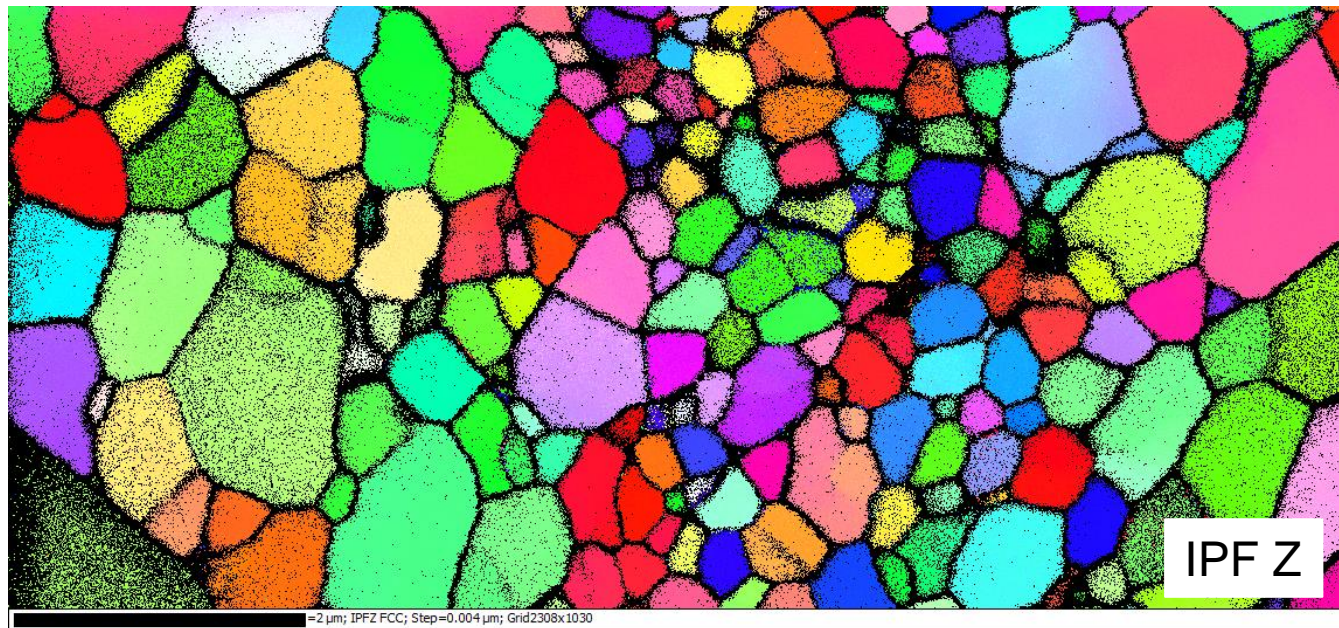


Pole figures show that regions across linear feature have similar orientations. Shear band cut across a grain.

EBSD of deformed Al-Si alloy – TKD from FIB prepared sample in linear band



Small
equiaxed,
recrystallized
grains in linear
feature.



TKD map acquired at
4 nm step size

Analysis of linear features in Al-Si

Linear features are consistent with ASB's because:

1. Evidence of deformation
2. Evidence of local heating – Si dissolved in local region
3. Evidence of recrystallization in shear band – TKD shows fine equiaxed grains in ASB region

EBSD characterization of whiskers offer many unique challenges:

Spatial resolution – whiskers have one dimension that is quite small

Geometry – Whiskers may cast shadows on detector screen that appear as bands adding indexing difficulties

We may want to know the whisker geometry with respect to the growth surface

Out-of-plane geometries are more difficult to deal with than planar (polished) samples.

Goal is to correlate crystallographic growth directions with physical growth angles



Sn whiskers can be reliability concern in electronic devices due to possibility of forming shorts.

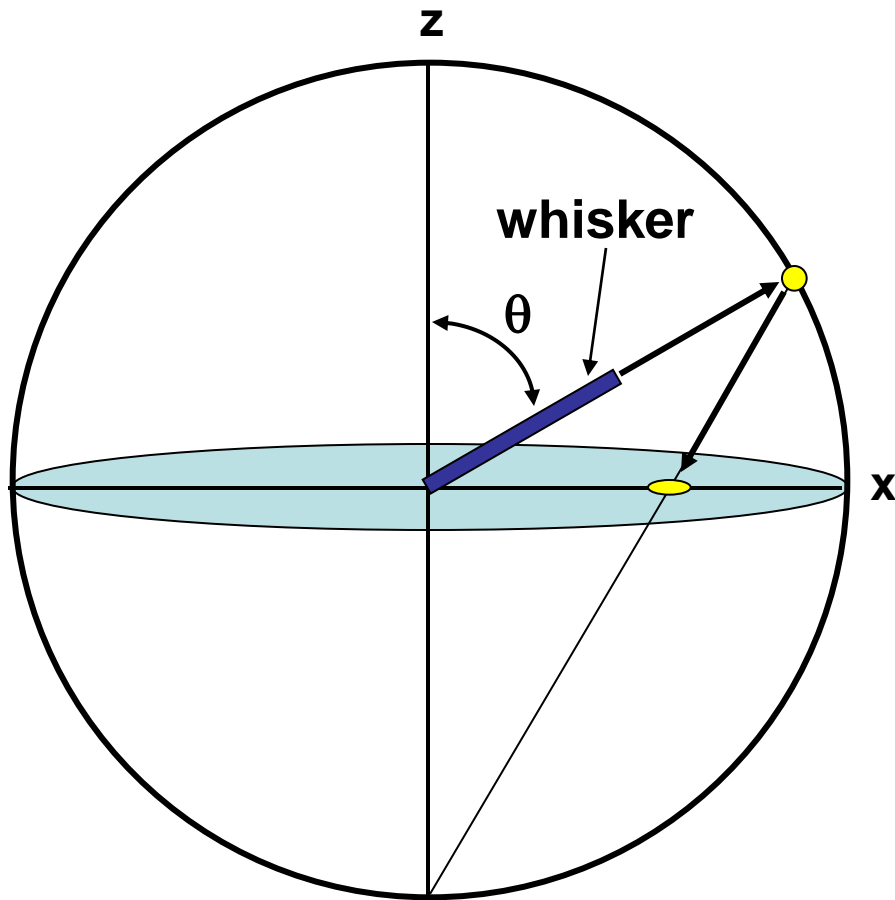
Whiskers “in-situ” aligned with tilt axis and independent measurement of growth angle

- 1. Image whiskers un-tilted and align axis with tilt axis of SEM stage. Measure projected length of whisker.**
- 2. Tilt sample to EBSD geometry and collect and index patterns from whisker. Measure projected height of whisker tip.**
- 3. Use parallax and geometry to determine whisker angle.**
- 4. Collect EBSD patterns and index. Mathematically rotate orientation matrix by measured whisker angle about Y axis to bring growth axis onto Z axis of pole figure.**
- 5. Plot inverse pole figure of Z-direction**

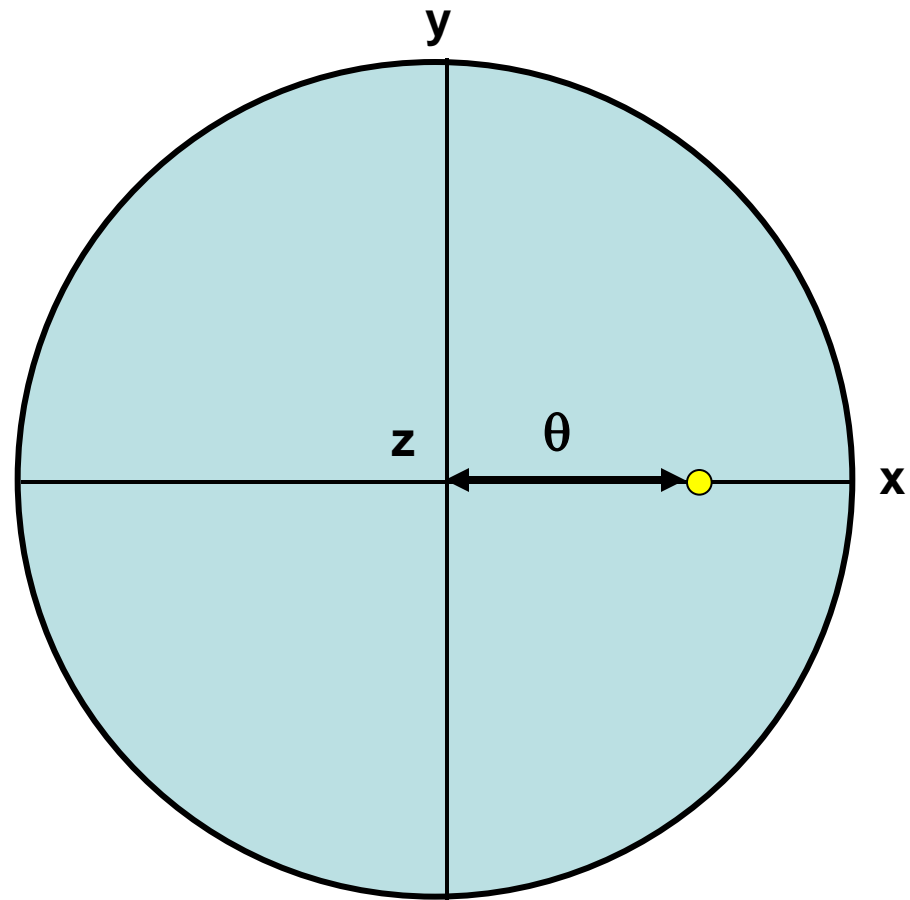
Advantages – retains whisker geometry, allows whisker axis to be unambiguously identified, independent measurement of whisker angle wrt sample surface, can use inverse pole figures for display

Disadvantages – neither fast or easy - about 40 whiskers per day

Whiskers “in-situ” aligned with tilt axis

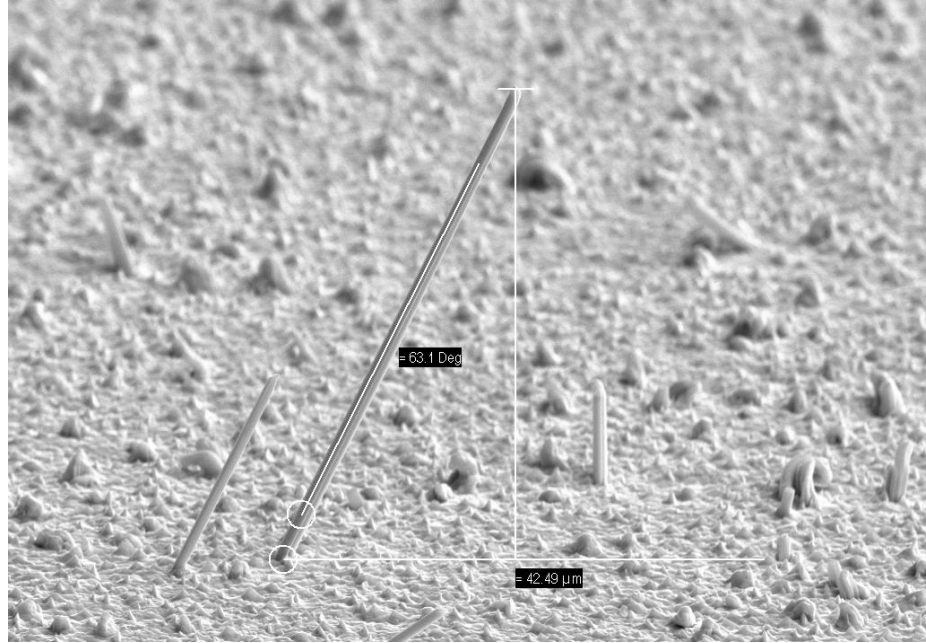
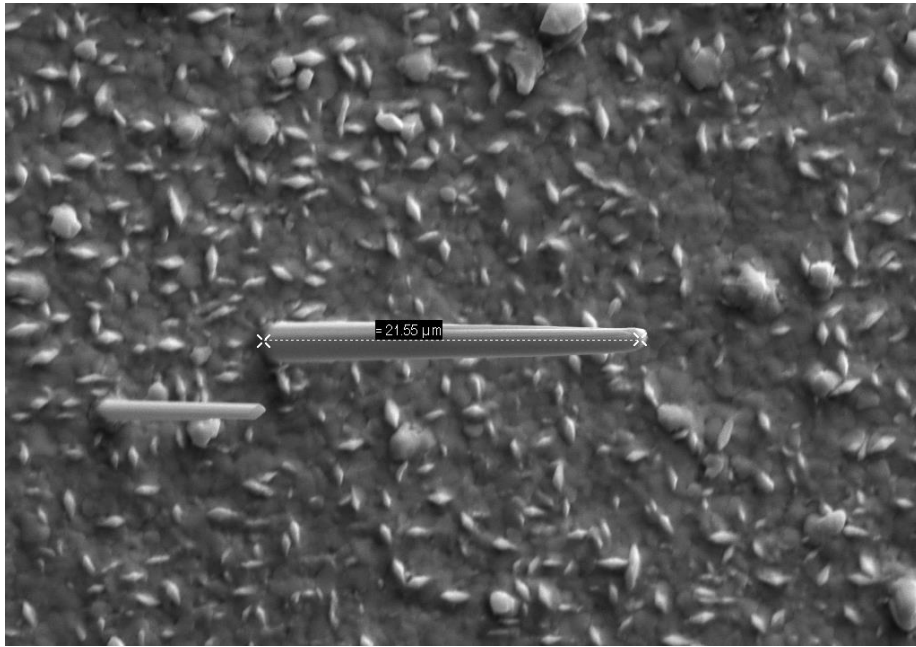


Whisker is in the x - z plane



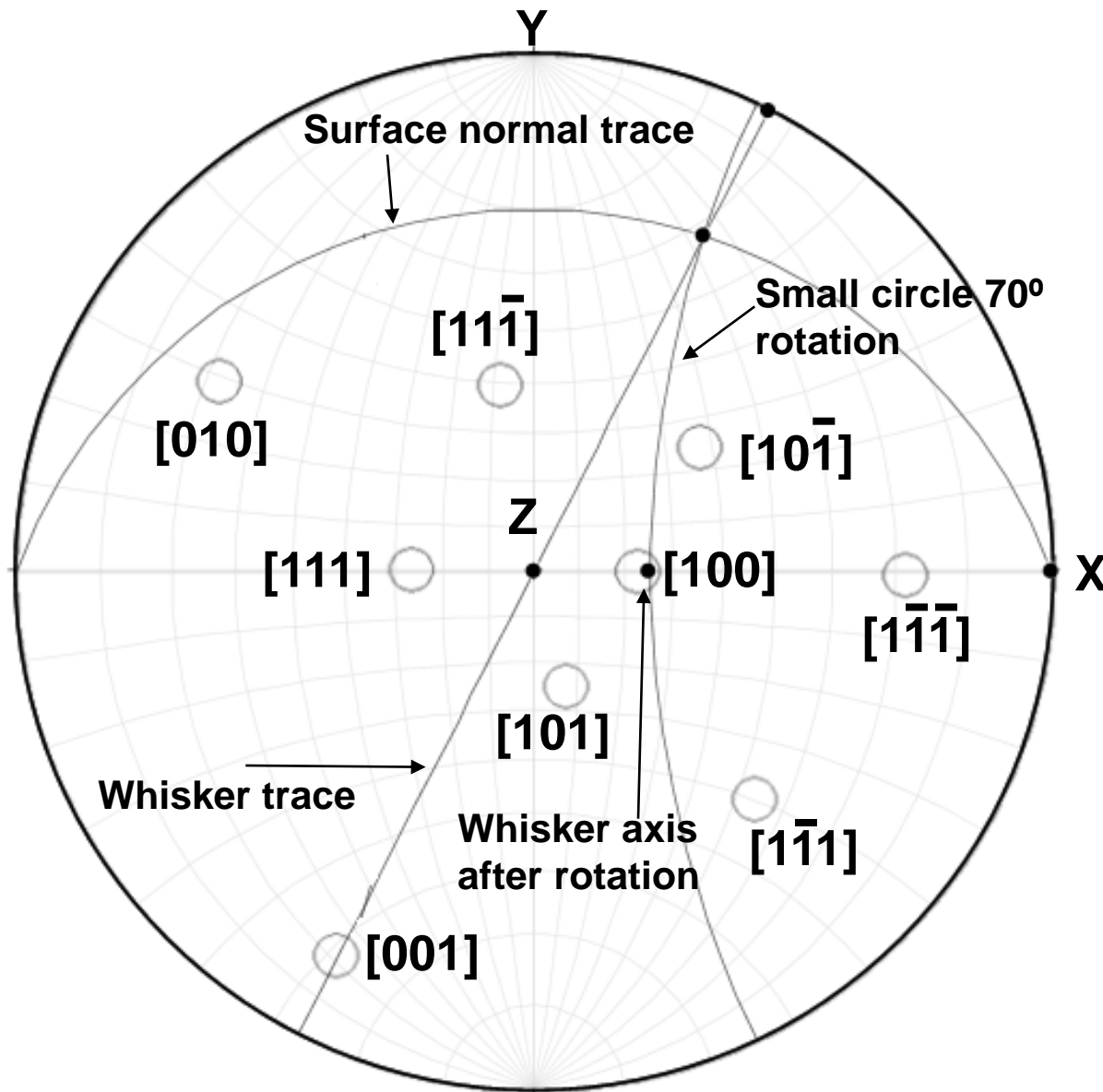
Pole figure or stereographic projection

Use Stereographic projections to determine whisker growth axis



Determine whisker normal trace from two views of sample. In this case with no tilt and with 70° tilt.

Use Stereographic projections to determine whisker growth axis



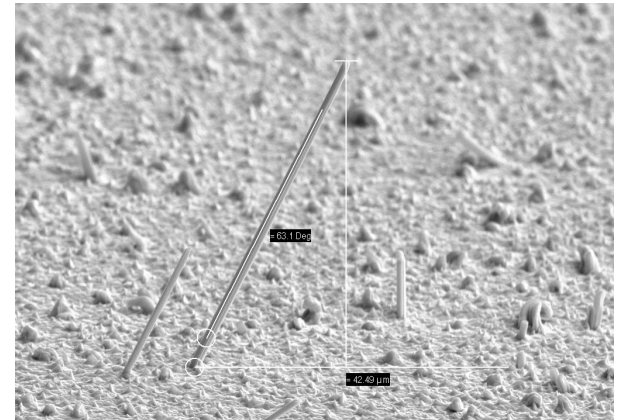
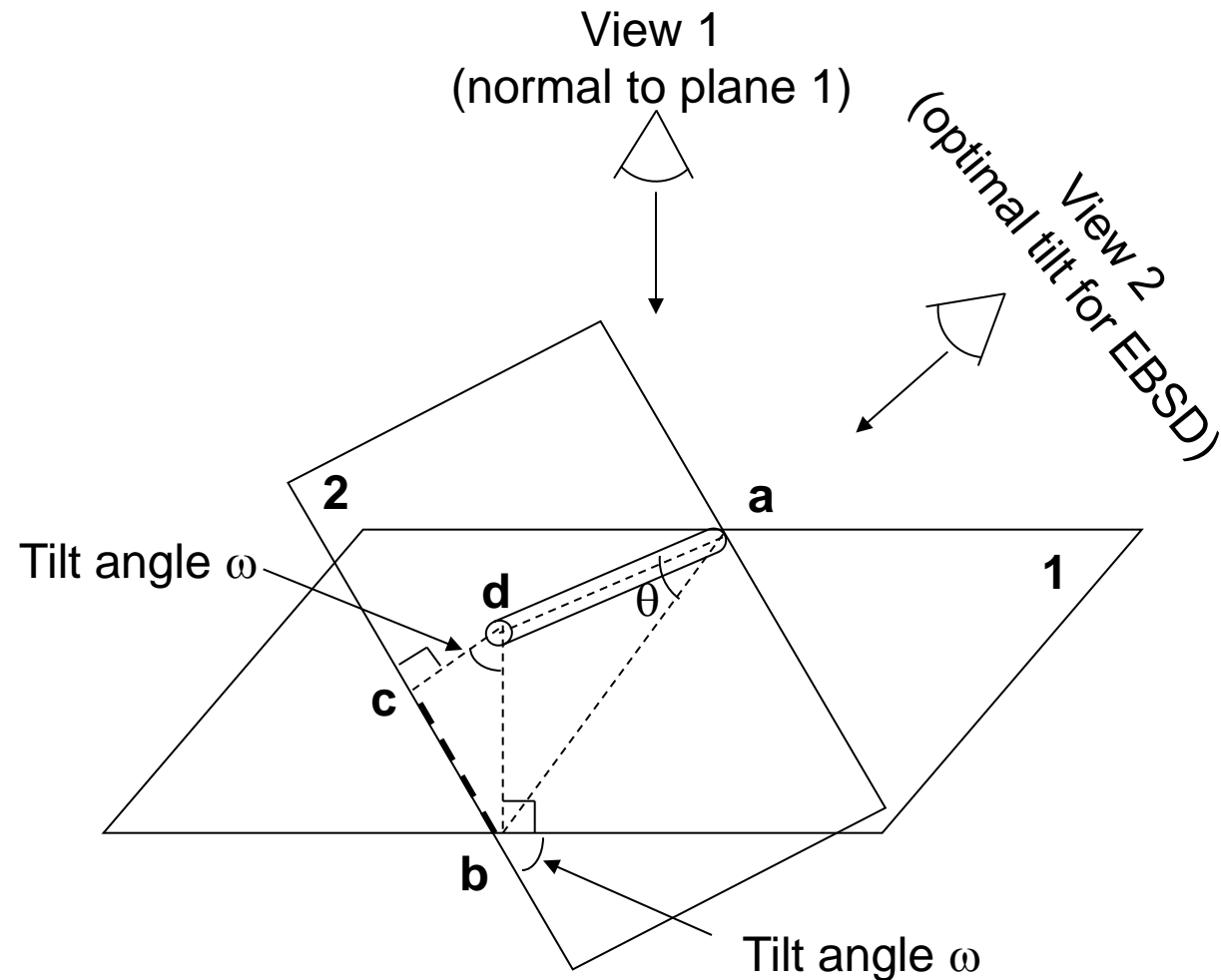
Surface normal plane contains whisker axis

Whisker axis is also contained in plane through the beam direction and inclined 63.1° to the x-axis

Draw great circles representing these planes and the intersection is the whisker axis.

Rotate axis (about tilt axis) along small circle -70° to get equivalent orientation to EBSD

Whiskers “in-situ” aligned with tilt axis



Tilt corrected height = measured height (at tilt)/ $\cos(90^\circ - \text{tilt angle})$

Tilt corrected height(db) = $42.5(cb)/\cos(90^\circ - 70^\circ) = 45.3 \mu\text{m}$

Whisker angle = $\text{ArcTan}(\text{tilt corrected height(db)}/\text{projected length(ab)})$

Whisker angle from surface = $\text{ArcTan}(45.3/21.5) = 64.6^\circ$

Orientation matrix rotation (for Oxford/HKL systems)

$$OM_{uvw} = \begin{bmatrix} a_1 & b_1 & c_1 \\ a_2 & b_2 & c_2 \\ a_3 & b_3 & c_3 \end{bmatrix} \quad \text{Where the columns represent the uvw with respect to x, y and z}$$

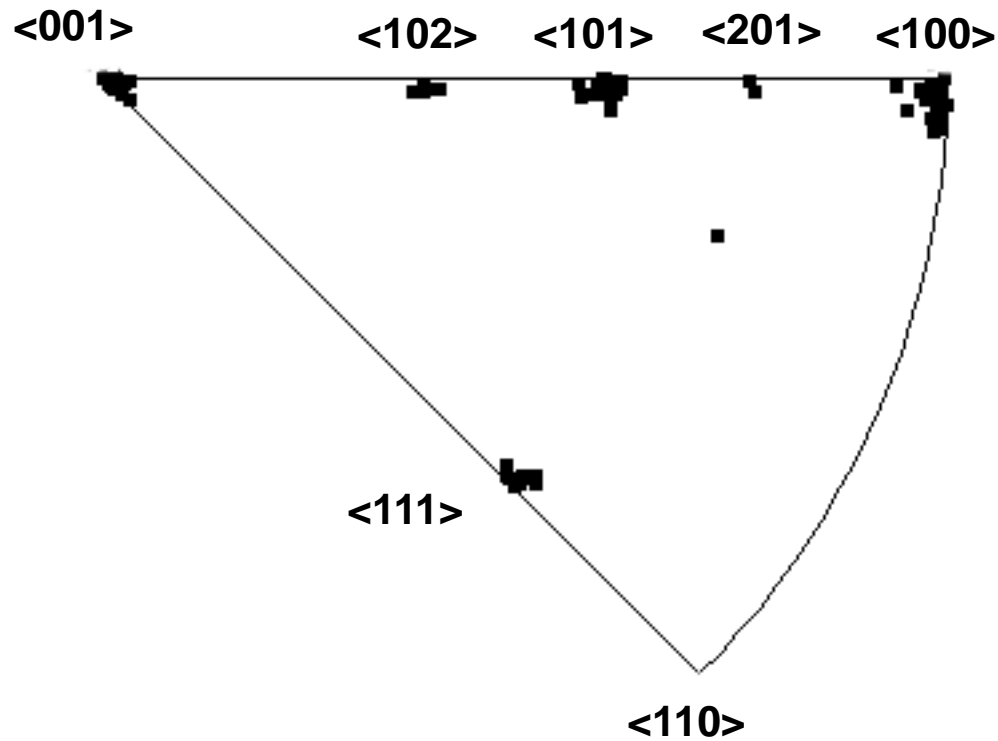
$$OM_{ortho} = \begin{bmatrix} 5.82 & 0 & 0 \\ 0 & 5.82 & 0 \\ 0 & 0 & 3.17 \end{bmatrix} \begin{bmatrix} a_1 & b_1 & c_1 \\ a_2 & b_2 & c_2 \\ a_3 & b_3 & c_3 \end{bmatrix} \quad \text{Multiply OM by the transformation matrix for Tin to get Cartesian coordinates}$$

$$OM_{ROT} = OM_{ortho} \begin{bmatrix} \cos \theta & 0 & -\sin \theta \\ 0 & 1 & 0 \\ \sin \theta & 0 & \cos \theta \end{bmatrix} \quad \text{Rotate OM about y-axis, angle is between sample normal and whisker axis.}$$

$$OM_{final} = \begin{bmatrix} 0.1718 & 0 & 0 \\ 0 & 0.1718 & 0 \\ 0 & 0 & 0.3155 \end{bmatrix} OM_{ROT} \quad \text{Multiply rotated OM by the inverse metric tensor for Tin so that columns of } OM_{final} \text{ are UVW with respect to x, y and z.}$$

If we get this correct the growth axis is aligned with the z axis!

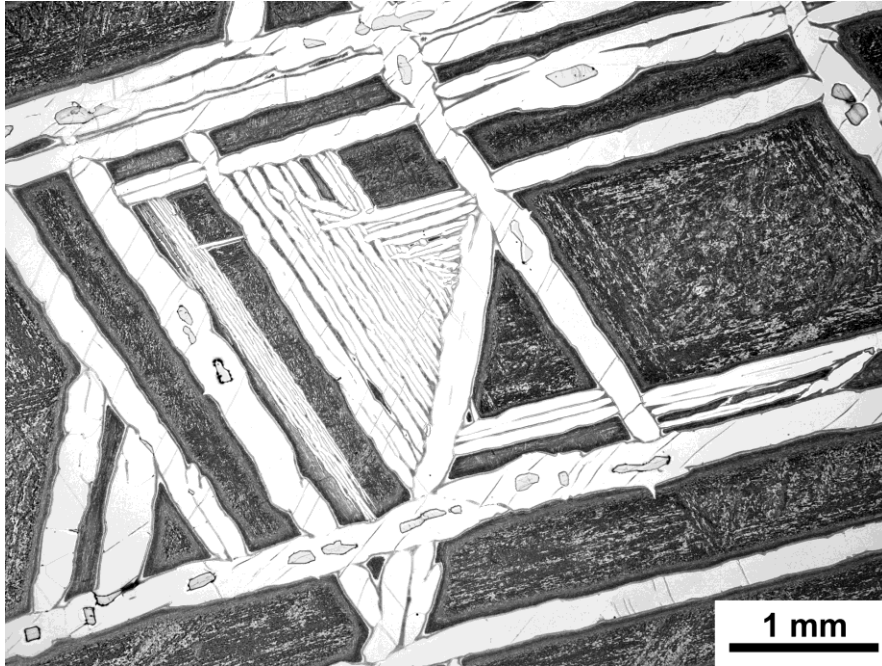
Characterization of 102 whiskers in-situ on three samples



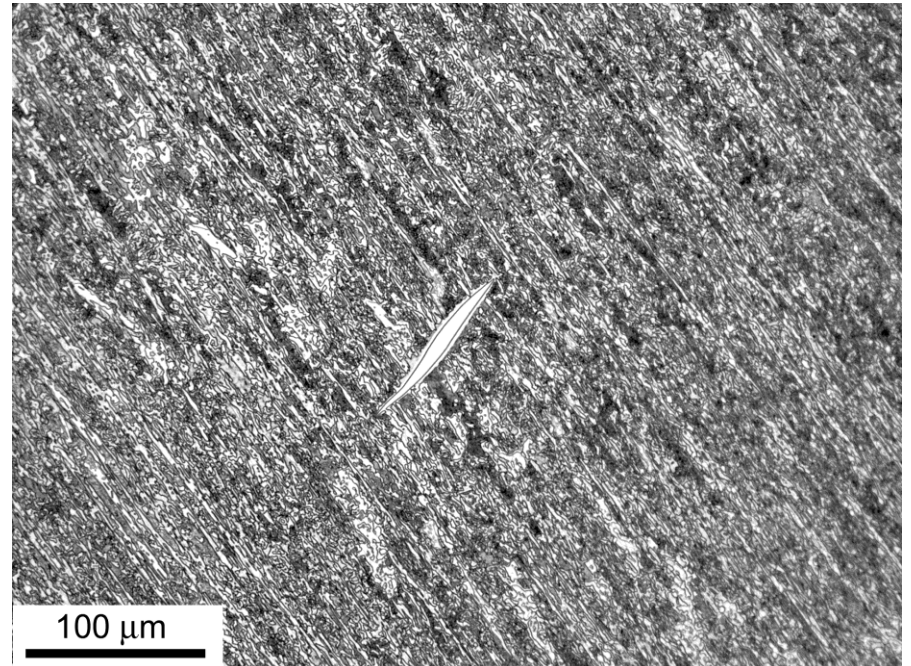
Distribution of growth axes for Sn Whiskers

$\langle 001 \rangle$	45
$\langle 010 \rangle$	19
$\langle 101 \rangle$	18
$\langle 111 \rangle$	11
$\langle 102 \rangle$	6
$\langle 201 \rangle$	2

EBSD's contribution to our understanding of the origins of the universe



Carlton

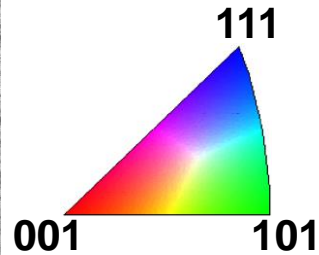
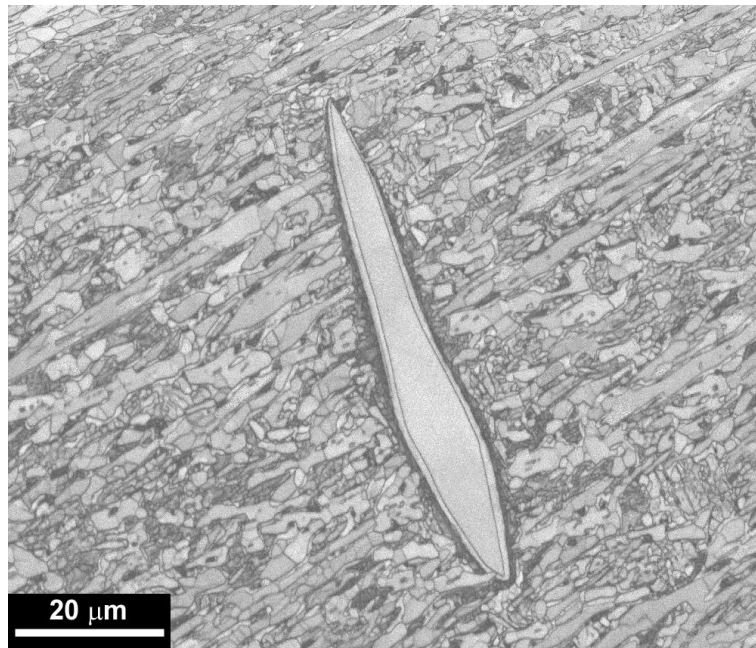


Cape of Good Hope

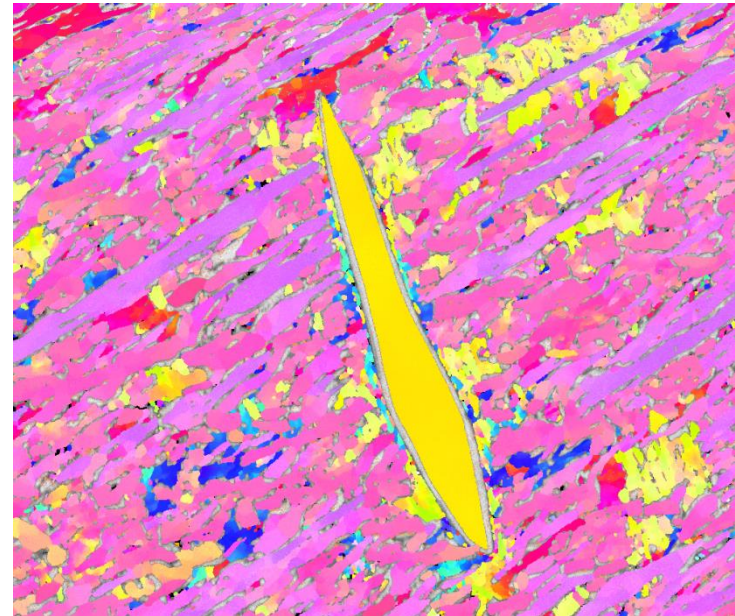
Microstructures of IVb Meteorites consist of Widmanstätten ferrite plus plesite (two phase mixture of austenite (taenite) and ferrite (kamacite)).

Microstructural studies can help develop an understanding of the cooling history of the meteorite.

Orientation mapping of iron meteorites



bcc

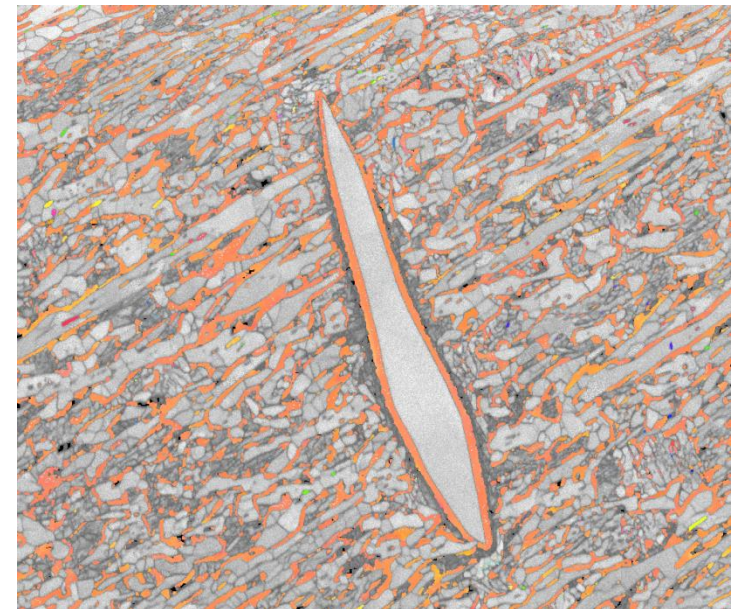


We can easily identify the fcc from the bcc phases of iron using EBSD.

All fcc has the same orientation!

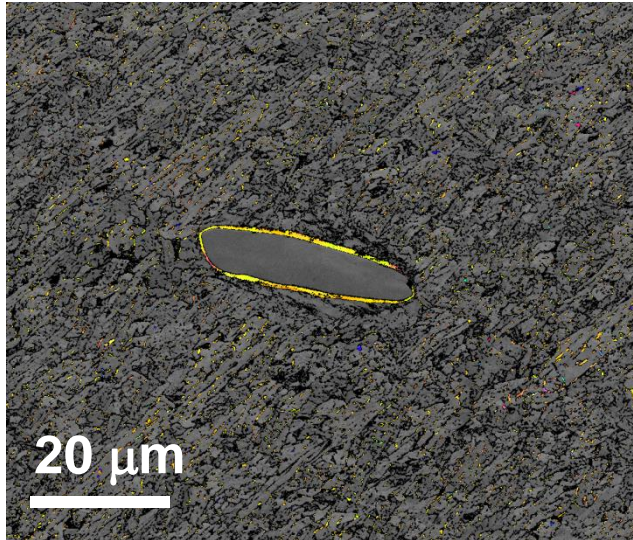
Appears to be retained austenite from a martensitic transformation

fcc

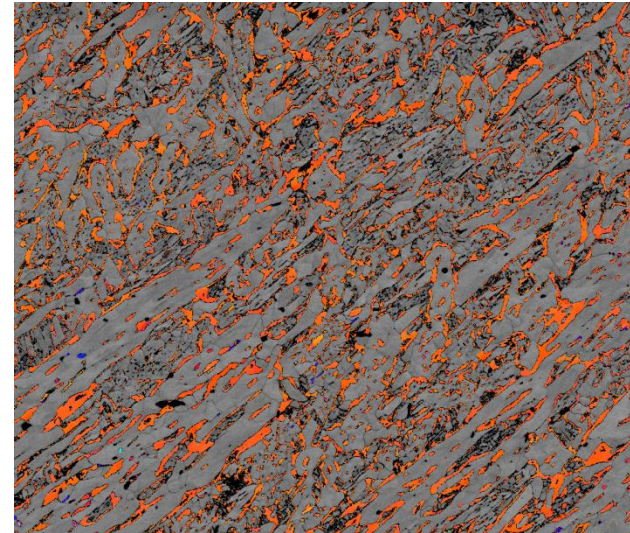


Plessite in IVb Meteorites – IPF maps of austenite phase

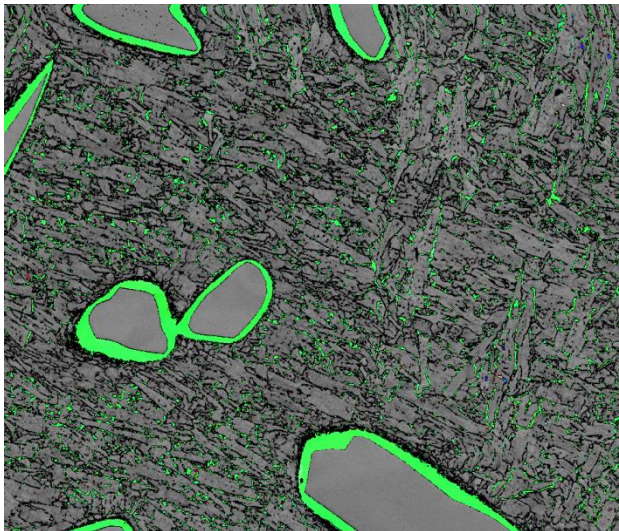
Chinga



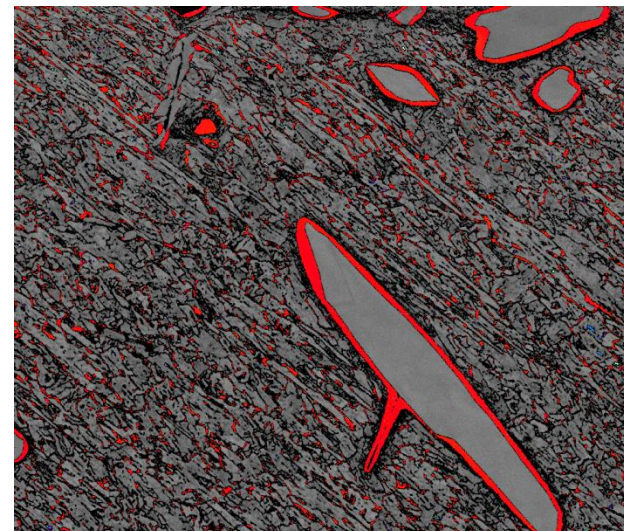
Hoba



Tawallah Valley

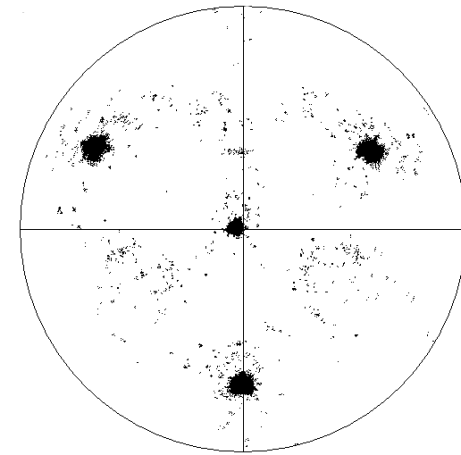
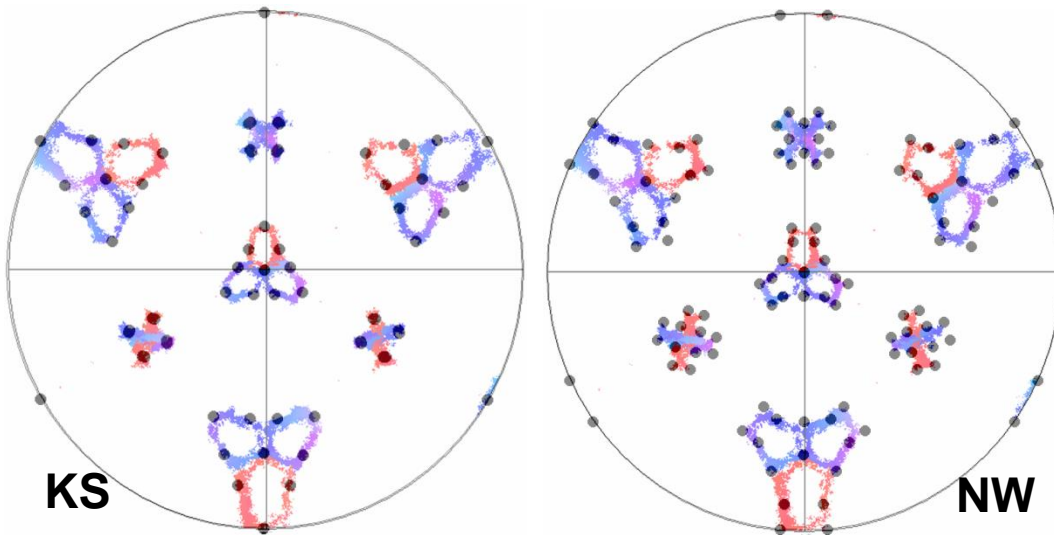


Weaver Mountain

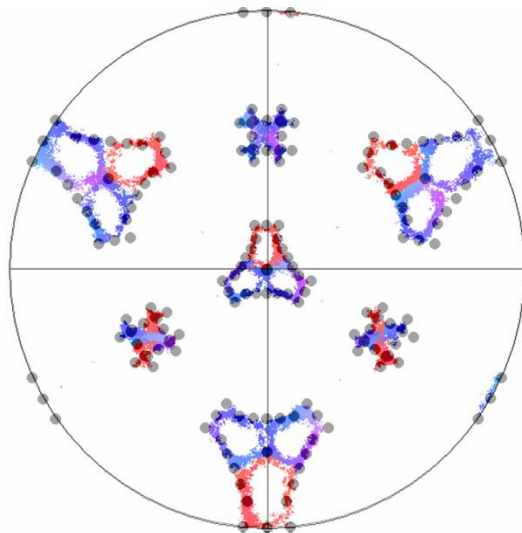


Note that in all cases all taenite (austenite) in a meteorite has the same orientation

Orientation relationships between Ferrite and austenite



**111
fcc Pole figure**



**KS +NW
bcc <110>Pole figure**

Complex $\langle 110 \rangle$ bcc pole figures indicate the orientation relationship between the fcc and the bcc phases is given by:

$$\langle 110 \rangle_{\gamma} \parallel \langle 111 \rangle_{\alpha} \quad \text{or} \quad \langle 011 \rangle_{\gamma} \parallel \langle 001 \rangle_{\alpha}$$

$$(111)_{\gamma} \parallel (110)_{\alpha} \quad (111)_{\gamma} \parallel (110)_{\alpha}$$

Kurdjumov-Sachs

Nishiyama-Wasserman

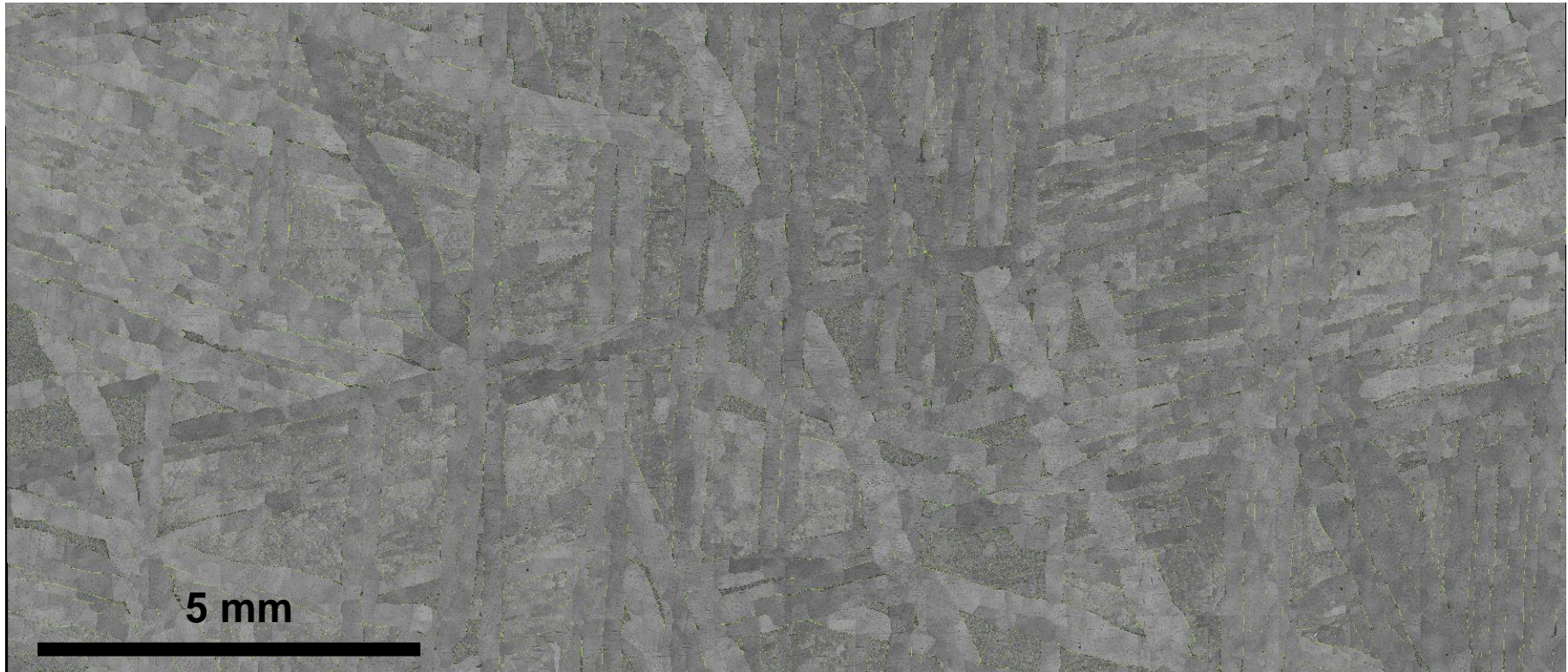
Or some combination of KS + NW

Large area mapping of Gibeon confirms the previous results



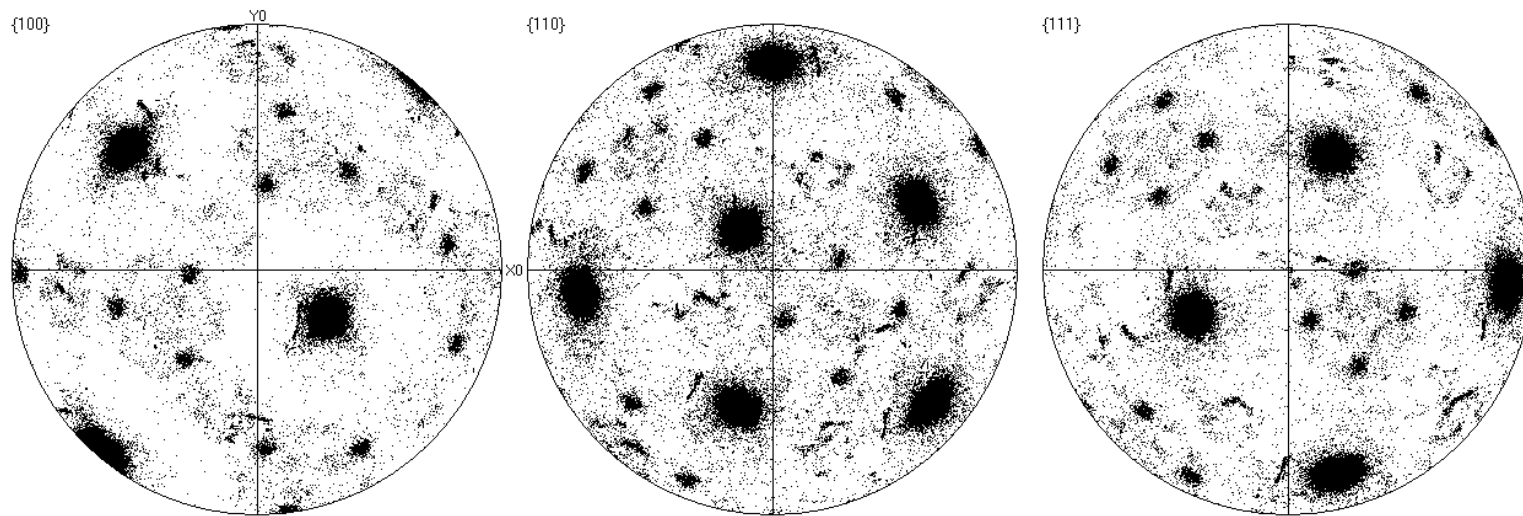
Polished section of Gibeon mapped over large area using $3\text{ }\mu\text{m}$ steps. Large Widmanstatten ferrite plates formed during slow cooling.

Large area mapping of Gibeon confirms the previous results

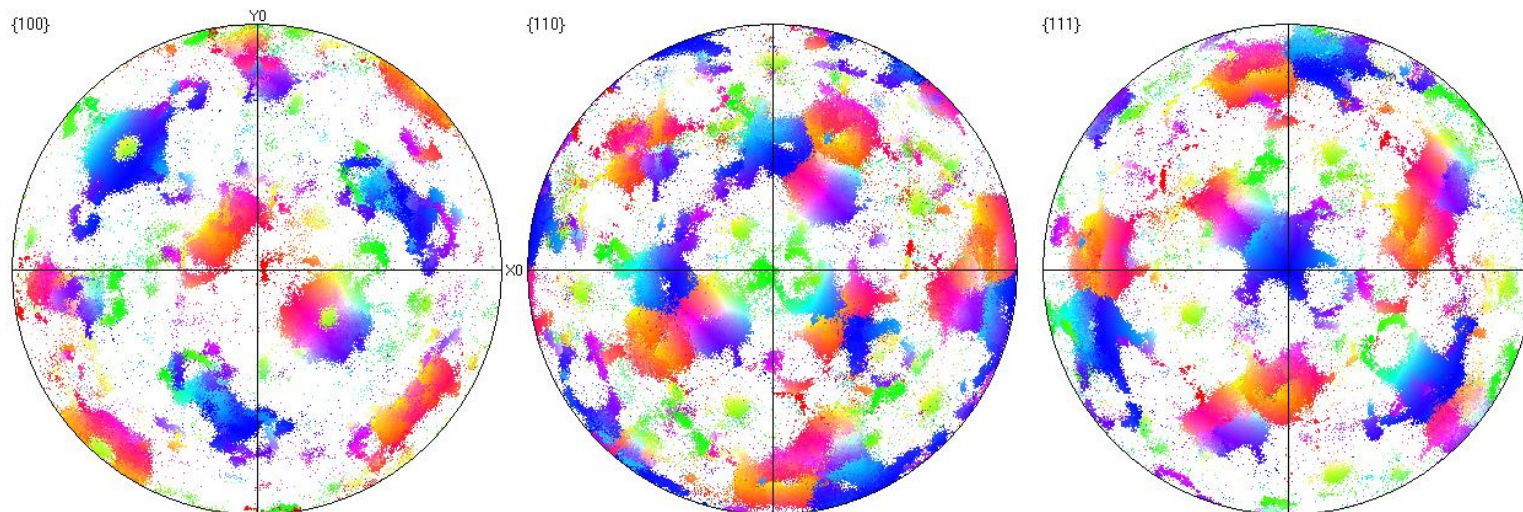


Along the Widmanstätten plates remains prior high temperature austenite all of the same orientation. This must be residual austenite from a grain that was larger than 22 mm across.

Pole figures of the Austenite and ferrite in the Gibeon meteorite



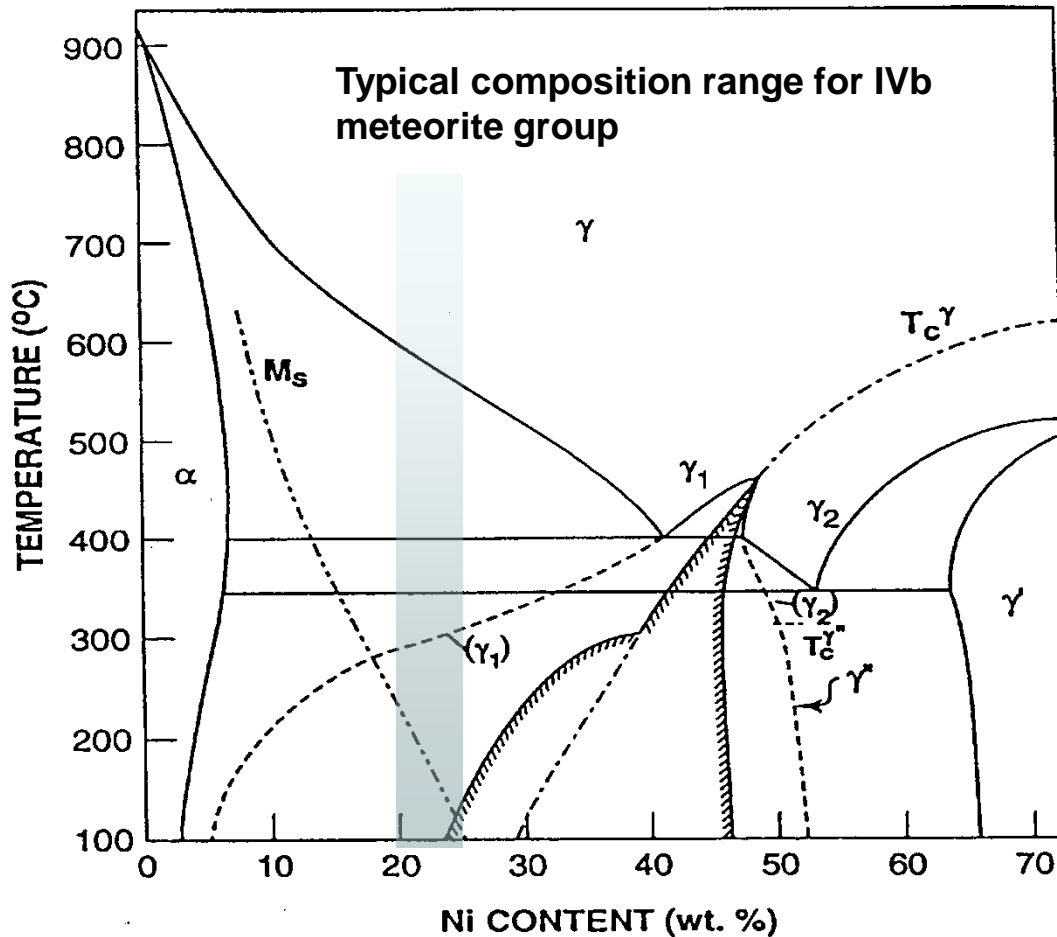
Pole Figures
[Gibeon_02 Specimen 2 Area Iron fcc (m3m) Complete data set 432658 data points Equal Area projection Upper hemispheres



Pole Figures
[Gibeon_02 Specimen 2 Area Iron bcc (old) (m3m) Complete data set 22608713 data points Equal Area projection Upper hemispheres

Consistent with results from smaller areas.

Iron-Nickel Phase diagram



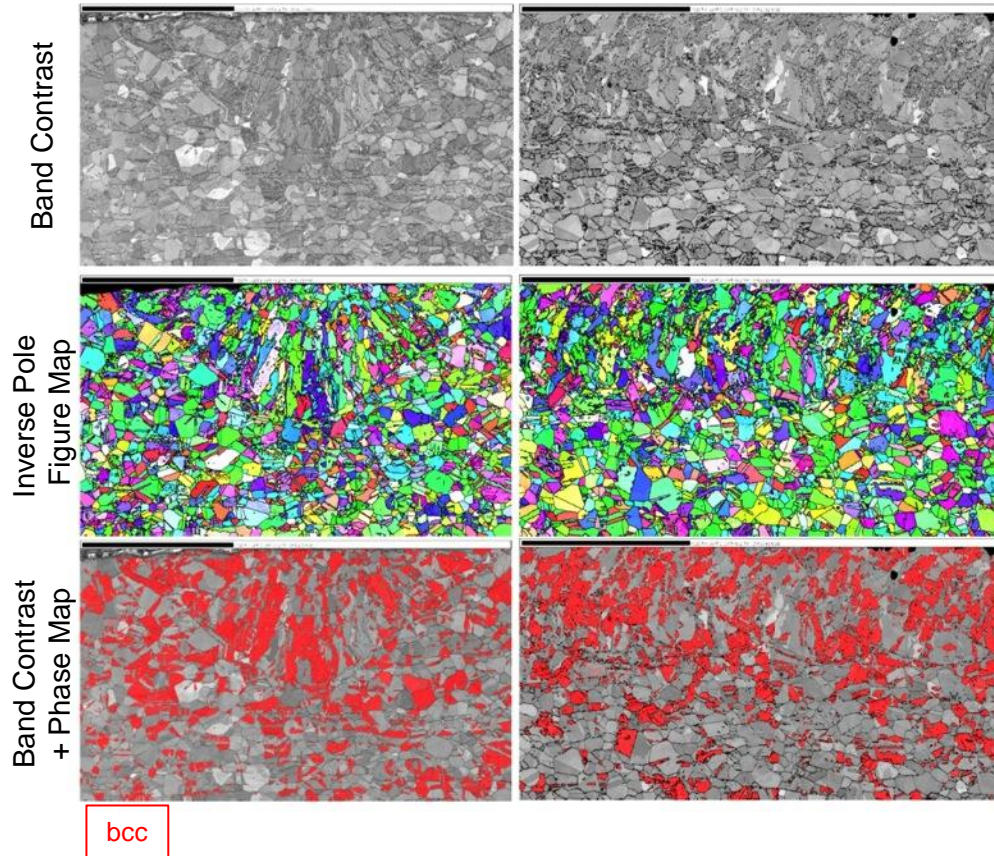
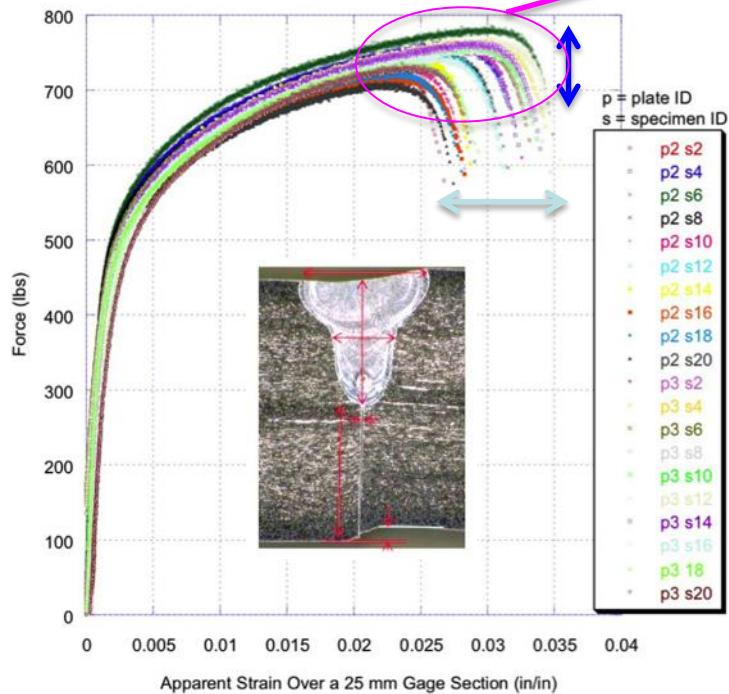
EBSD results gave a new insight to how plessite (a two phase region in meteorites containing both kamacite (bcc or ferrite to metallurgists) and taenite (fcc or austenite)).

$\gamma \rightarrow \alpha_2(\text{martensite}) \rightarrow \alpha + \gamma$ inconsistent with EBSD results

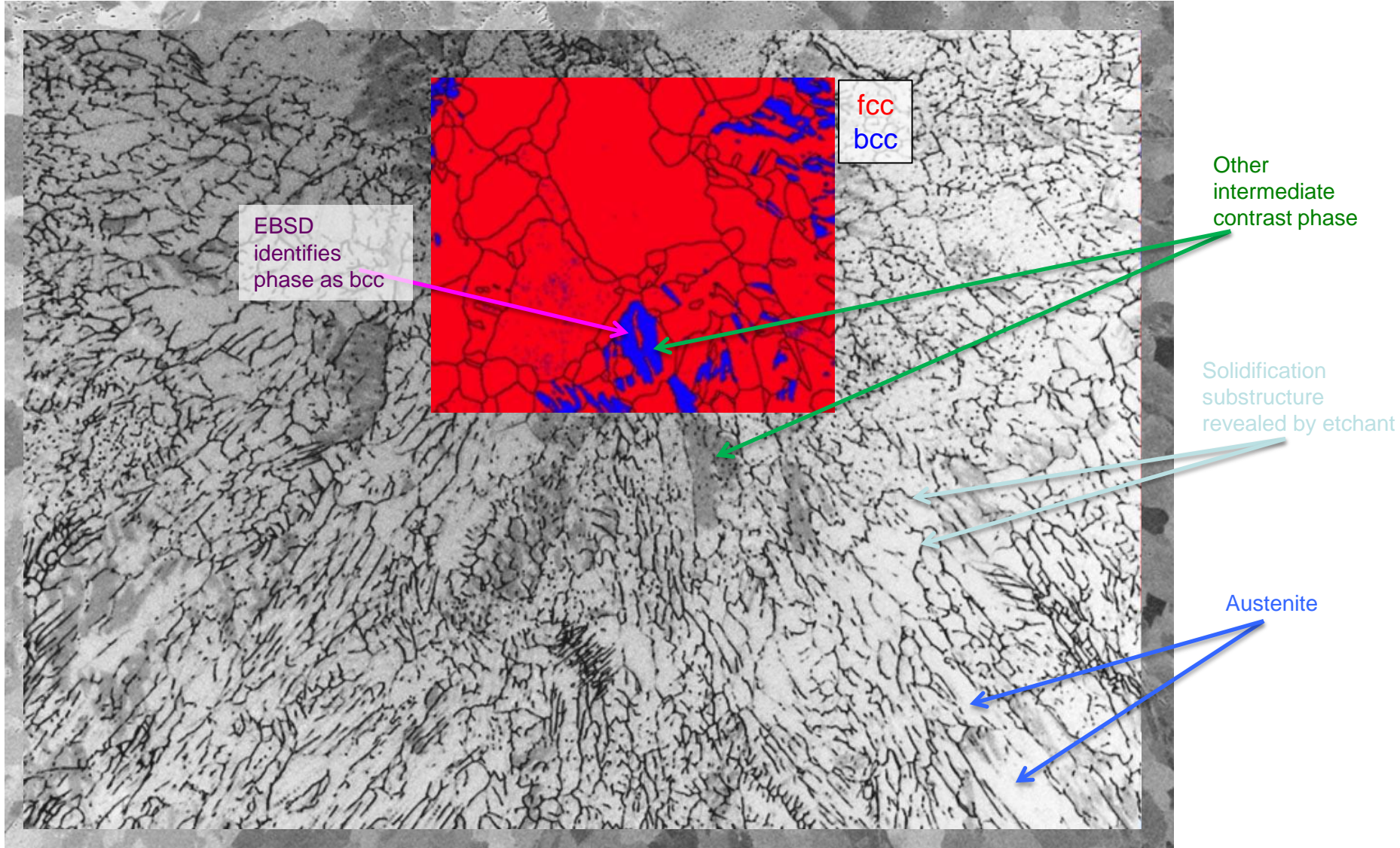
$\gamma \rightarrow \alpha_2(\text{martensite}) + \gamma \rightarrow \alpha + \gamma$ consistent with EBSD results

Understanding Austenitic Stainless Steel Weld Mechanical Behavior Variation Requires Accurate Knowledge of Phase Distribution

- 60-70% of mechanical behavior variation attributed to non-metallurgical factors (e.g., weld shape, joint geo., etc.)
- Characterization of property variation due to metallurgical factors requires accurate characterization of phase constitution



Metallographic Preparation for EBSD Phase Distribution Requires Additional Scrutiny



304L laser weld mechanically polished using conventional metallographic practices

Sample preparation possibilities – Electropolishing vs. mechanical methods

Standard metallographic procedure

**grind, polish, and finish with 3 hours on vibratory
polishing**

Electropolishing

Bath – 80% phosphoric 20% Butanol

Bath Temp - 70C

Applied current - 600mA/cm²

Polishing time – 2min

Di water rinse and N2 blow dry

**Note that electropolishing requires no preliminary surface
preparation and is really fast!**

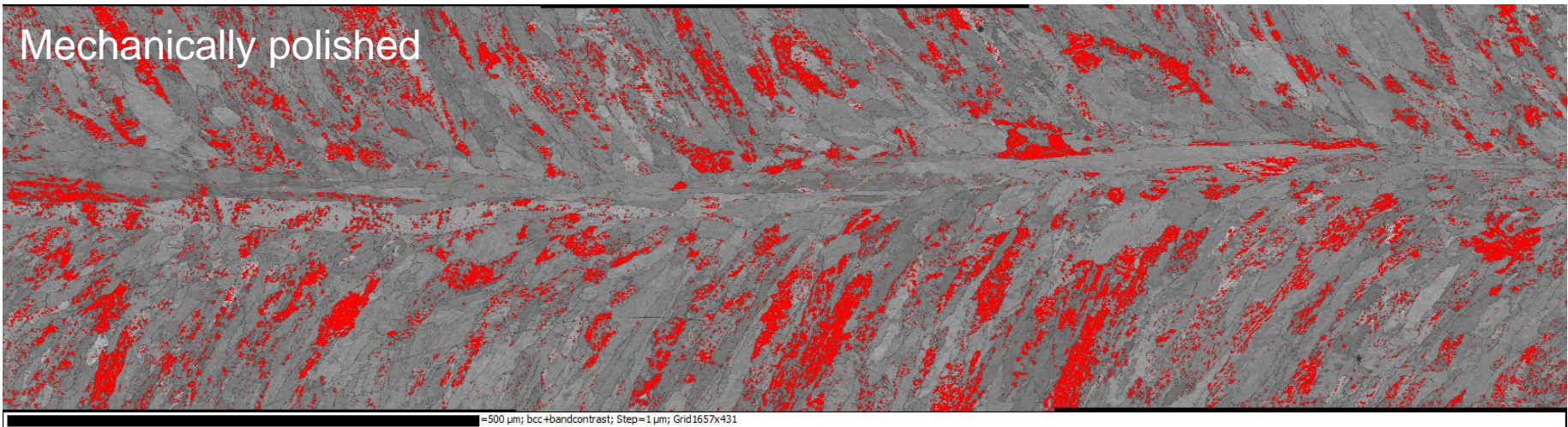
Bead on plate welds –Comparison of electropolished and metallographic polishing

Electropolished



Band contrast image with ferrite in red (0.1 area % ferrite)

Mechanically polished

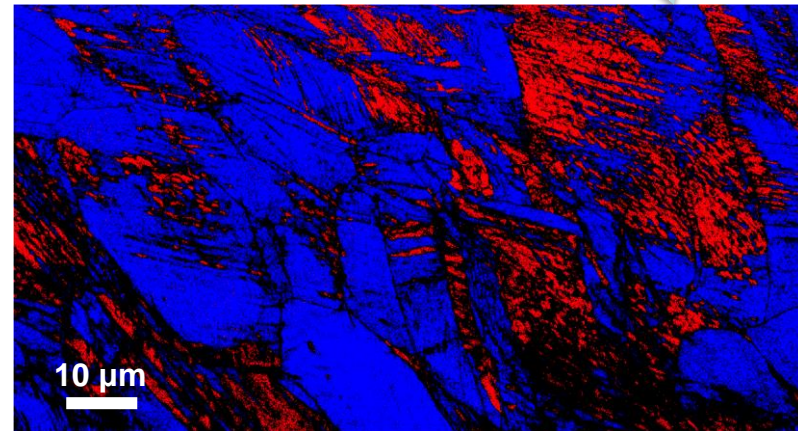
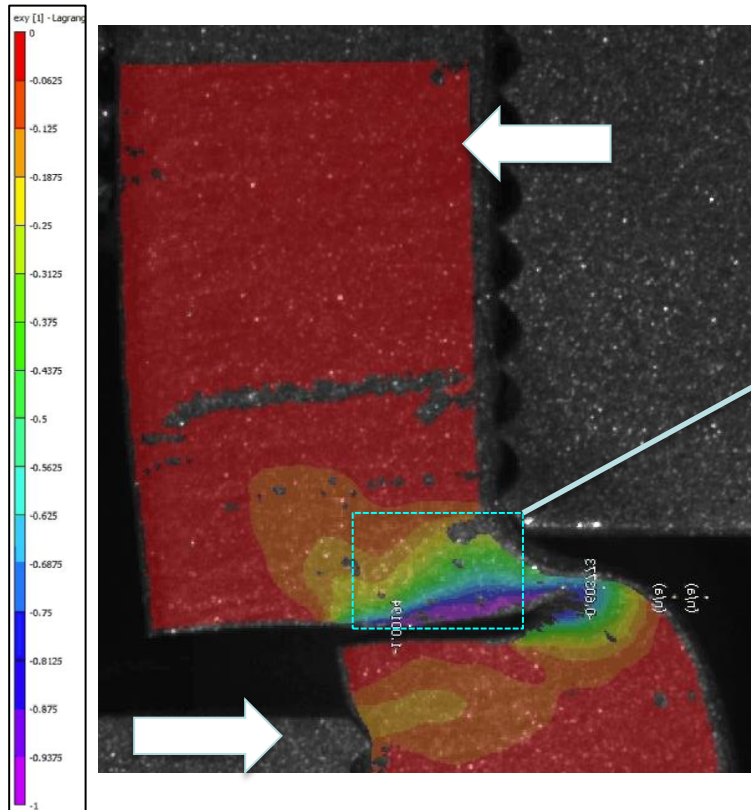
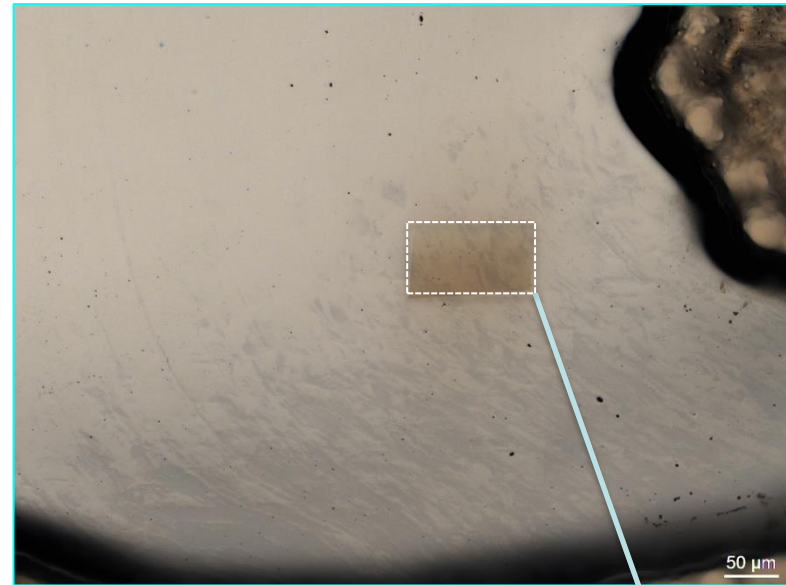


Band contrast image with ferrite in red (12 area % ferrite)!

304L Shear Test Specimens Exhibit Deformation-Induced Martensite

Controlled shear loading of commercial wrought 304L leads to room-temperature deformation-induced martensite

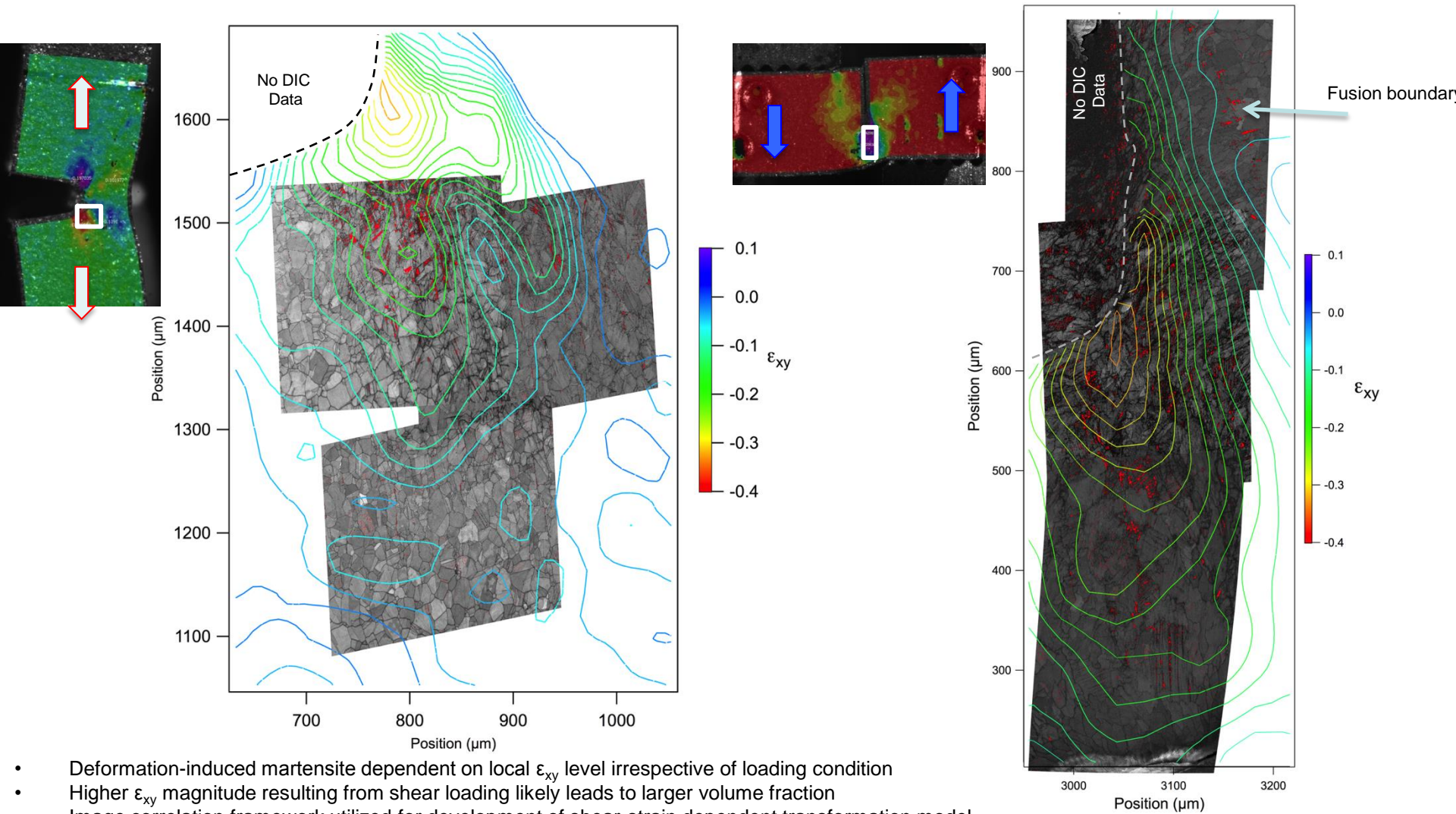
Optical micrograph of shear specimen (*electropolished*)



Phase Map: **fcc**; **bcc**

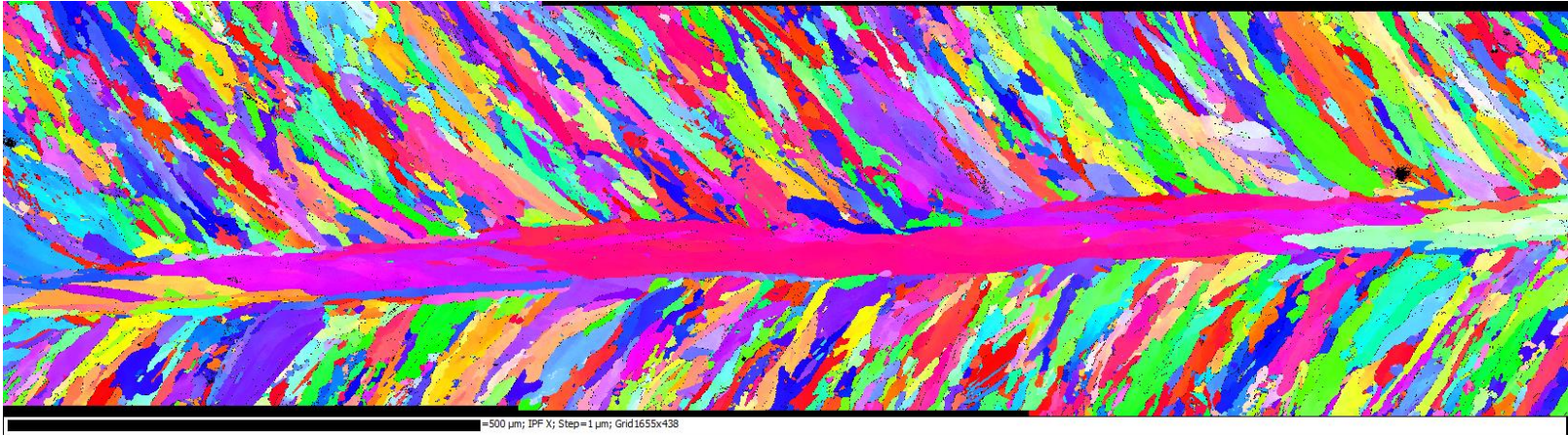
- Increasing Shear Strain
- Decreasing Diffraction Indexing Accuracy
- Increasing Martensite Fraction

Deformation-Induced Martensite Dependent on Local Shear Strain Level Irrespective of Loading Condition



- Deformation-induced martensite dependent on local ϵ_{xy} level irrespective of loading condition
- Higher ϵ_{xy} magnitude resulting from shear loading likely leads to larger volume fraction
- Image correlation framework utilized for development of shear-strain dependent transformation model

Unambiguous determination of ferrite content with EBSD in 304 steels



Mechanical polishing does not produce representative samples!

Unambiguous determination of ferrite content via EBSD requires electropolishing.

Shear strains can promote room temperature martensite in 304 stainless steels

Application of EBSD to understanding the structure of Additively Manufactured 304L

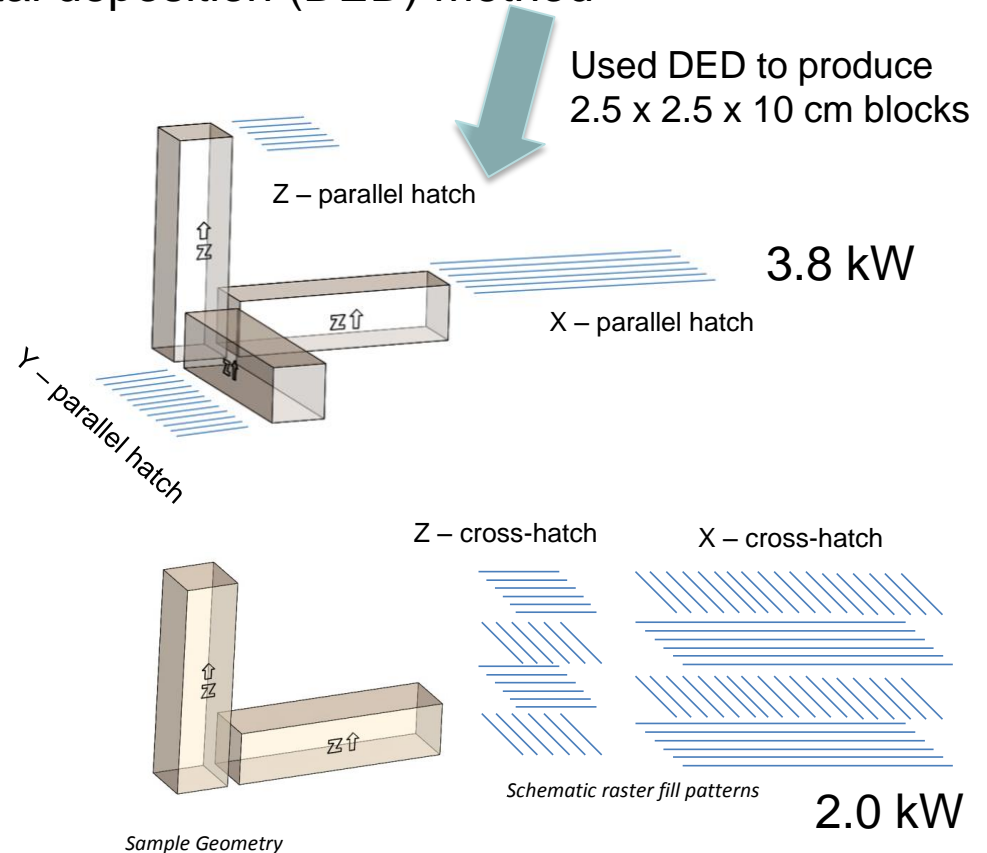
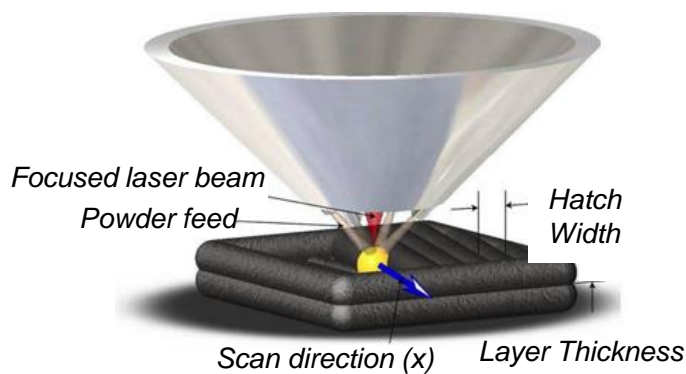
Understanding of solidification modes and microstructures needed as input to materials models.

Samples produced using direct metal deposition (DED) method

Application of EBSD to understanding the structure of Additively Manufactured 304L

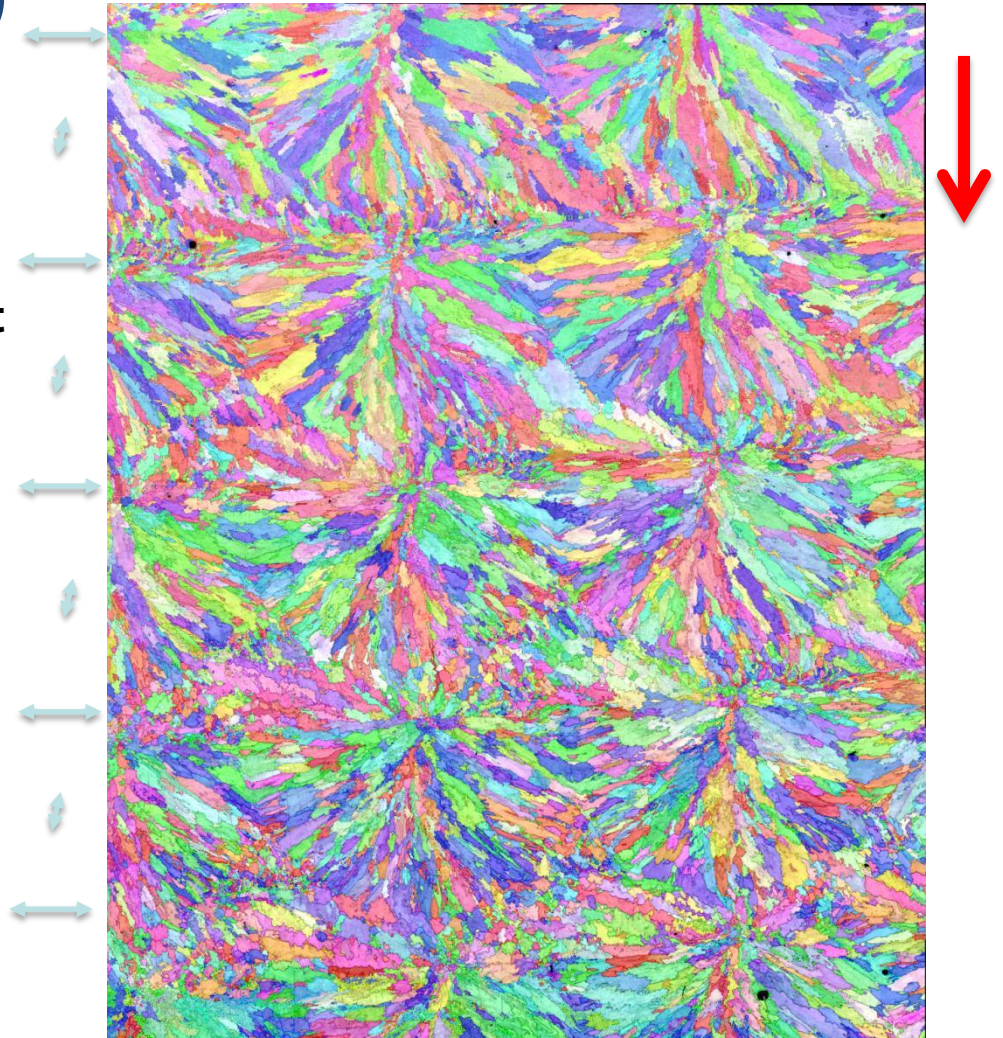
Understanding of solidification modes and microstructures needed as input to materials models.

Samples produced using direct metal deposition (DED) method



Large area views of microstructure of AM SS304L (2.0 kW)

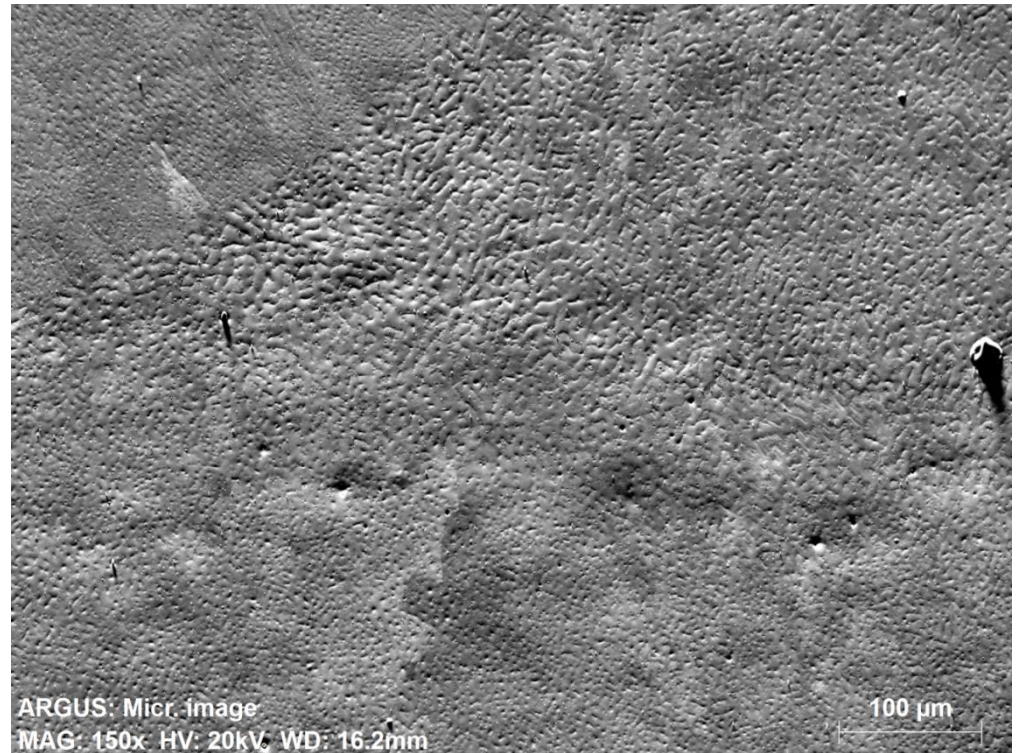
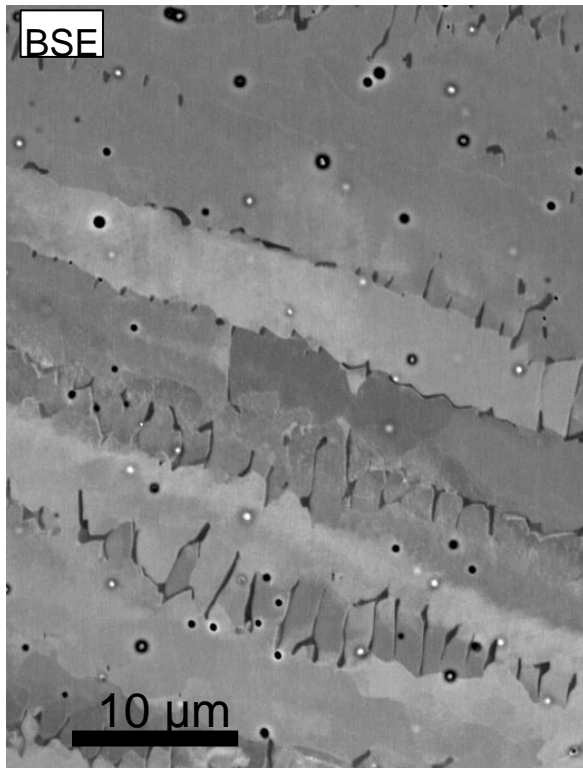
- Electron backscatter diffraction maps of electropolished surface.
- Example shown to right was built with a cross hatch approach.



IPF X + BC 6 mm wide by 10 mm high

Primary Ferrite or Primary Austenite Solidification?

As-deposited structure is a mix
austenite with fine ferrite precipitates.



Primary Ferrite or Primary Austenite Solidification?

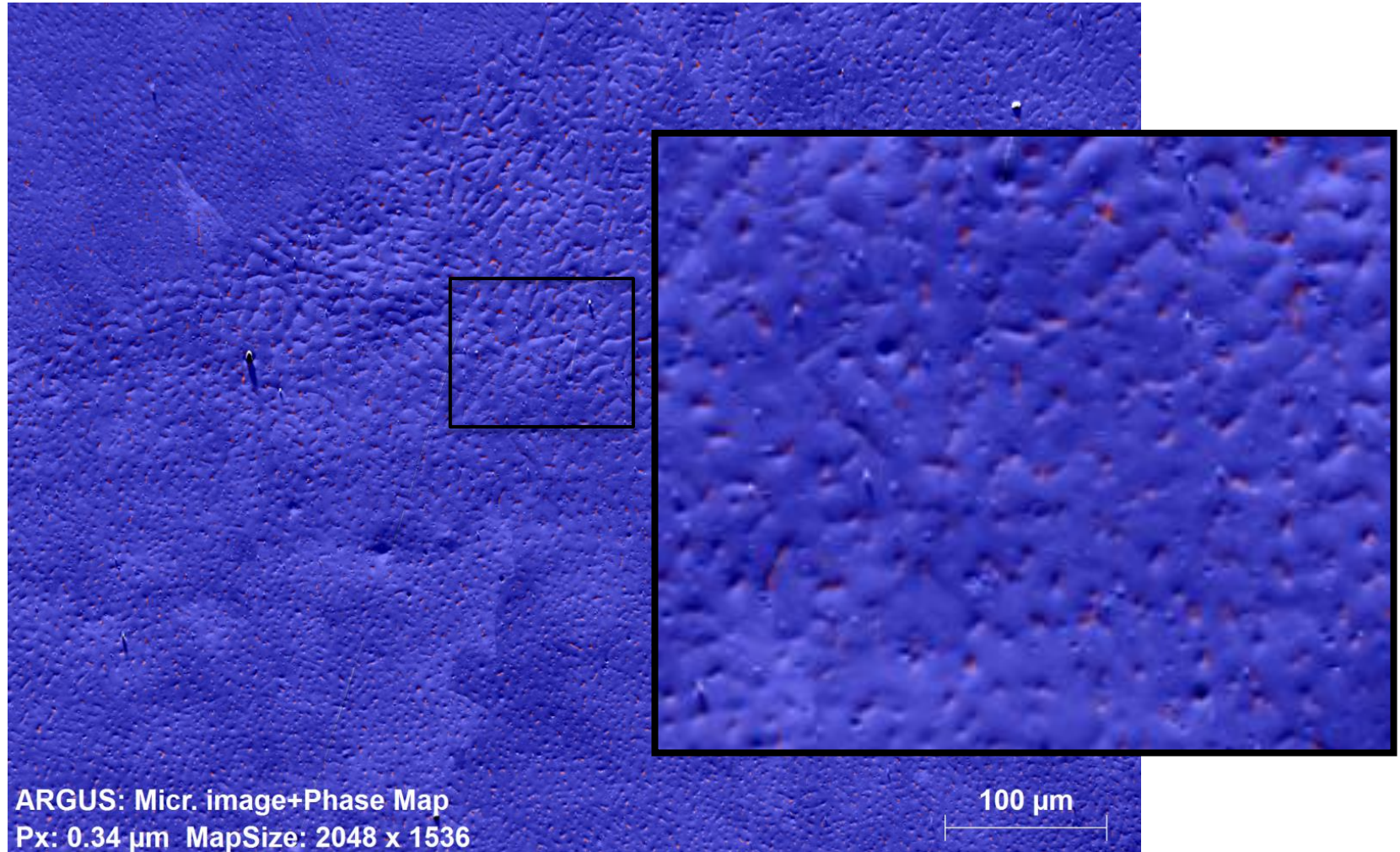
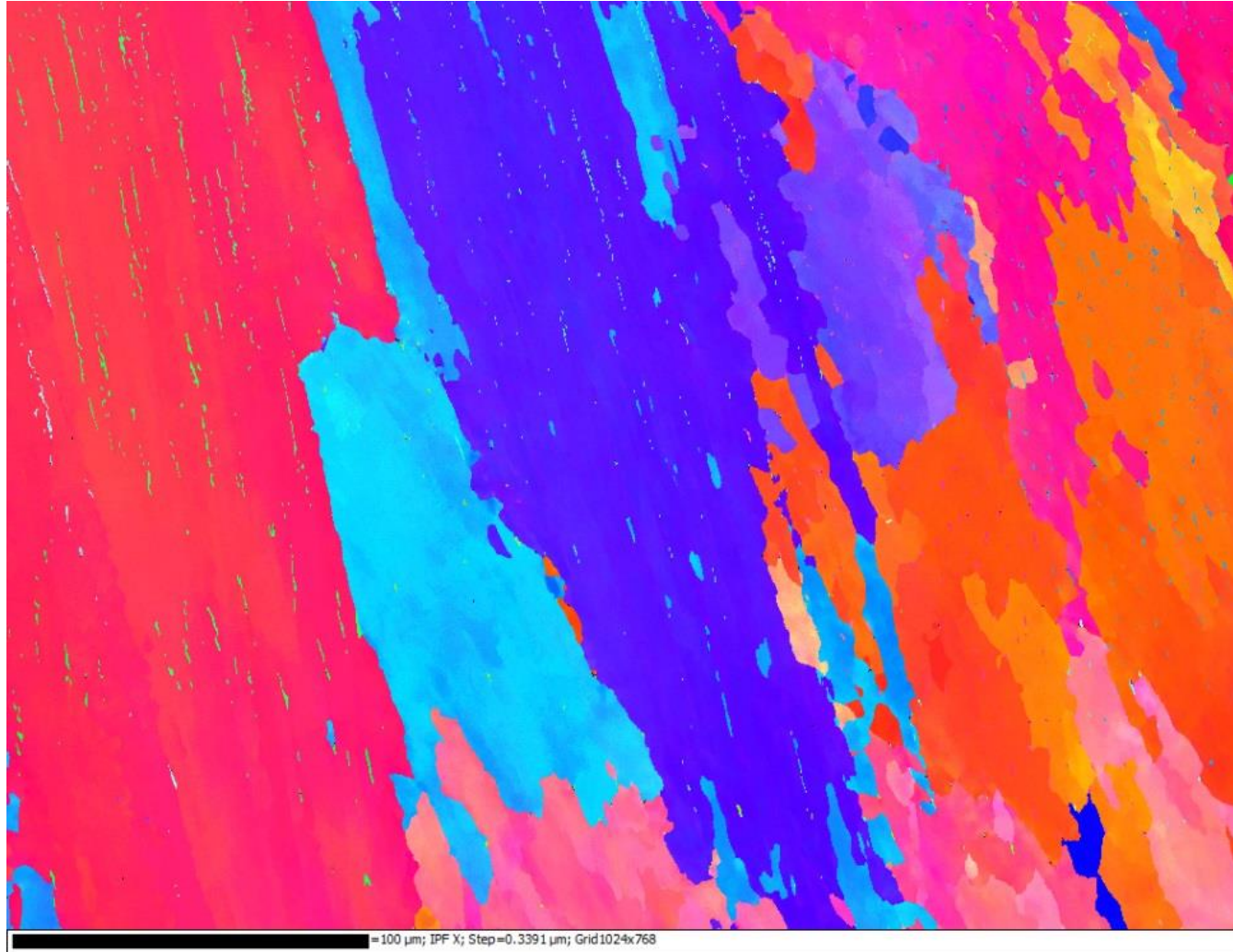


Image + austenite (blue)+ferrite (red)

EBSD reveals the fine ferrite precipitates in the austenite matrix.

Primary Ferrite or Primary Austenite Solidification?

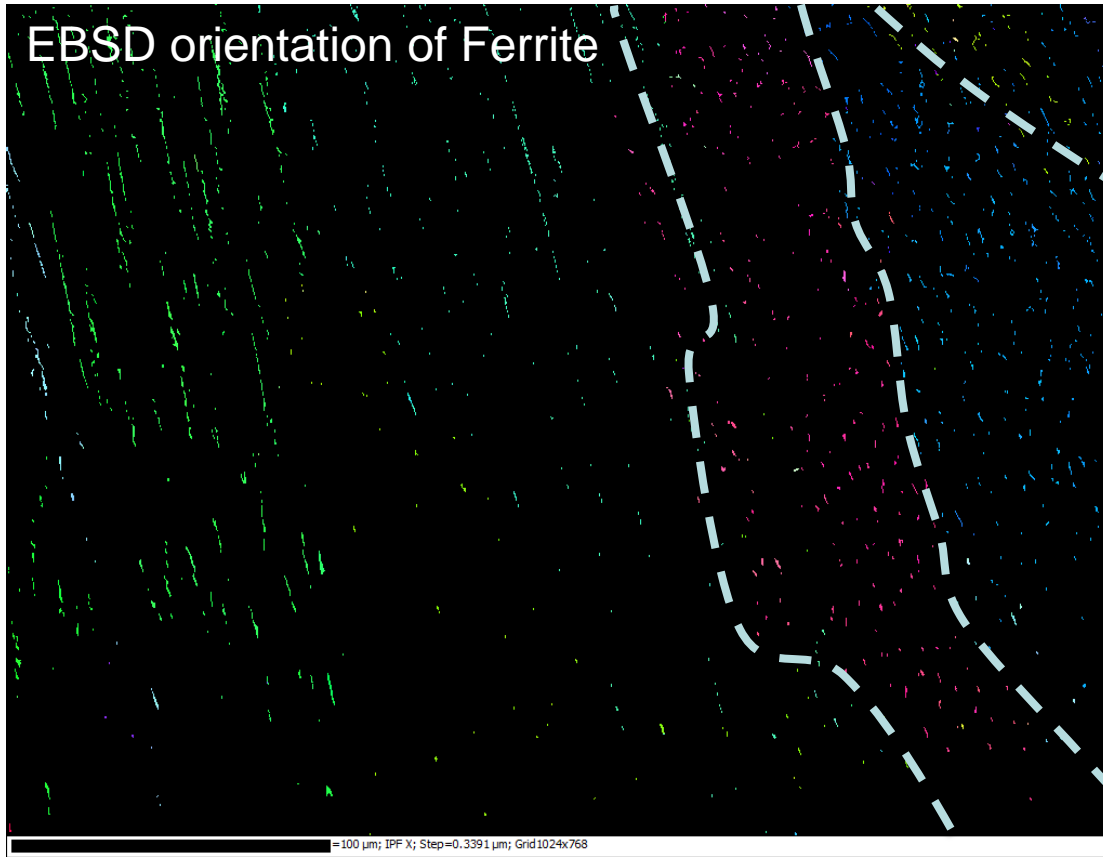
EBSD orientation of Austenite and Ferrite



Common Ferrite Orientation Within the Solidification Subgrains
Indicative of Primary Ferrite Solidification

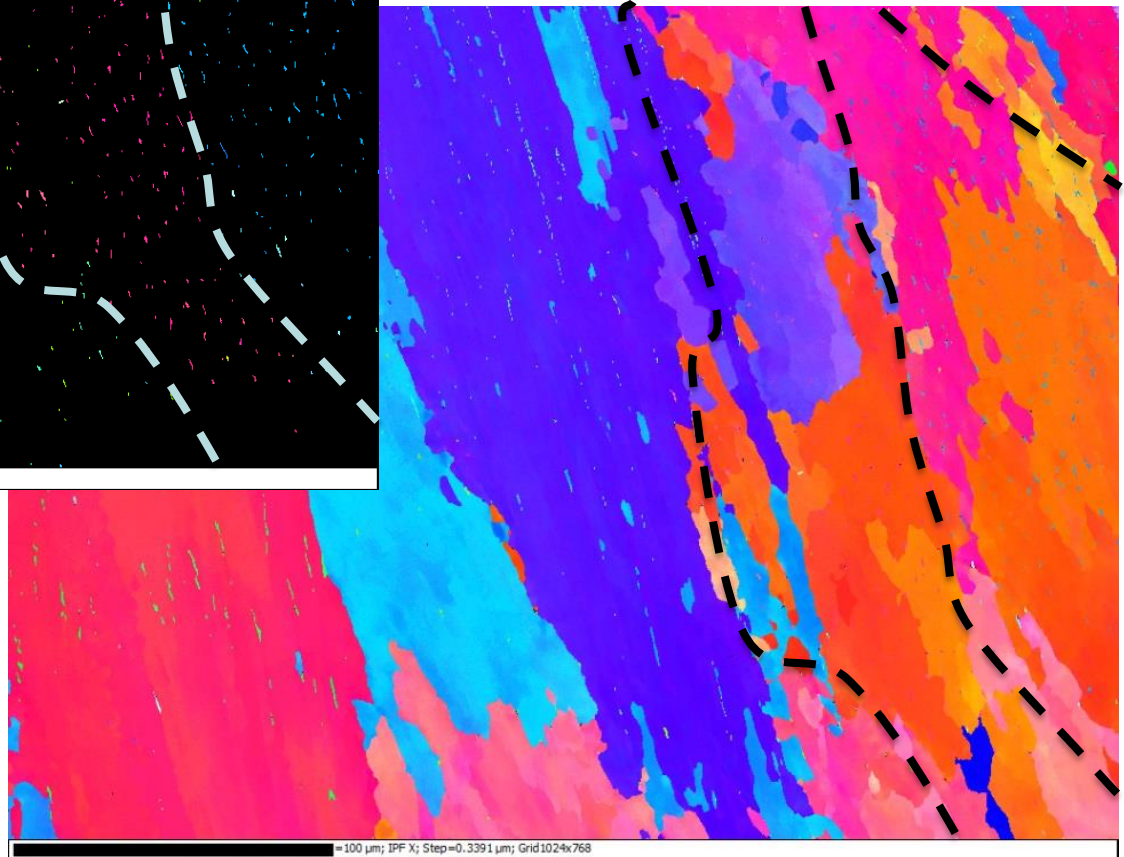
Primary Ferrite or Primary Austenite Solidification?

EBSD orientation of Ferrite

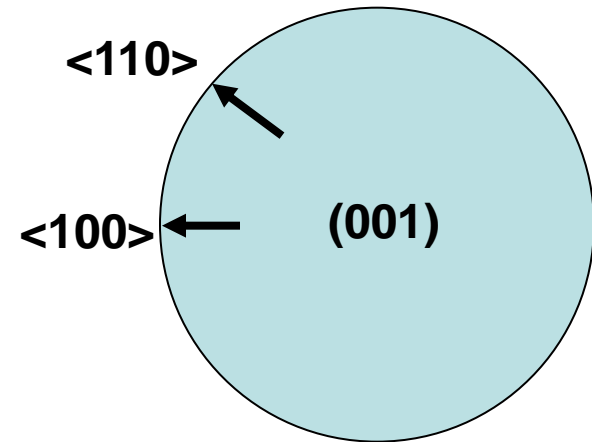
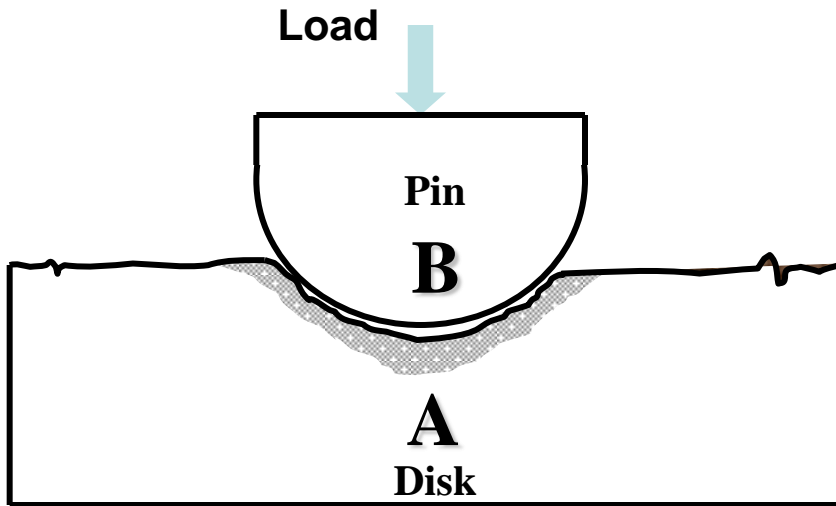


This is primary ferrite solidification and can be used to inform microstructural development models.

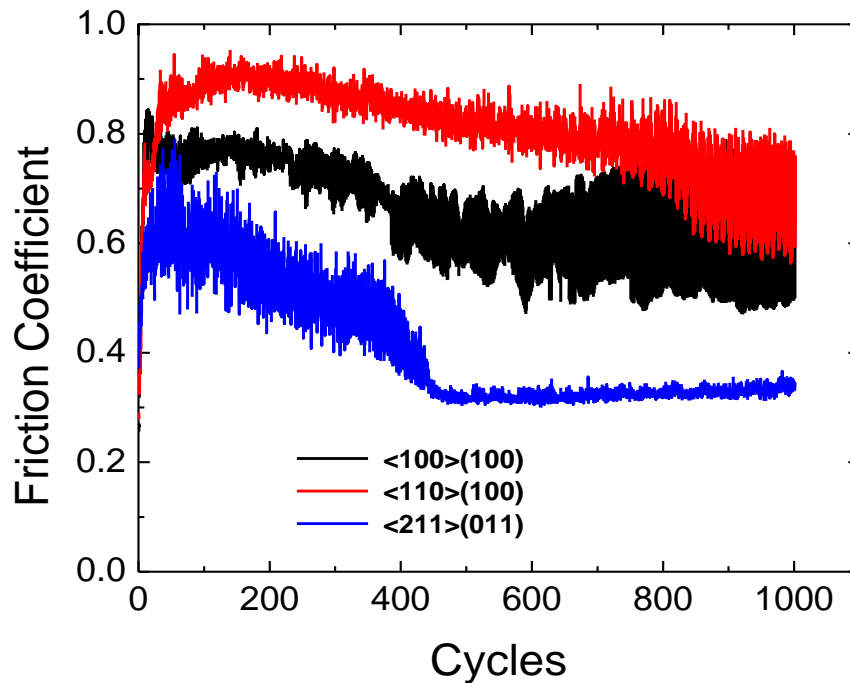
Common ferrite orientation allows prior ferrite grains (high temperature) to be visualized.



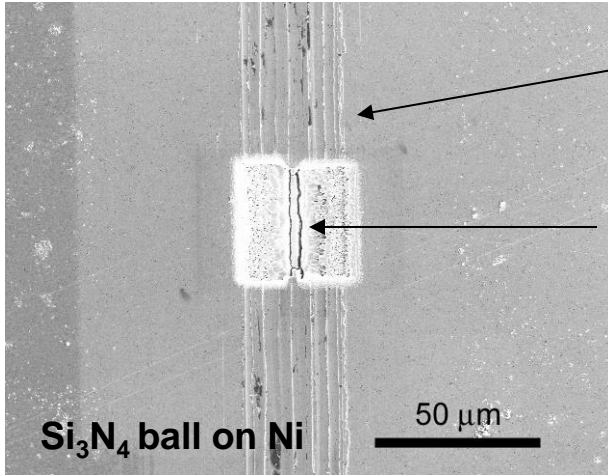
Understanding of friction using EBSD



Use EBSD in FIB to orient sample and then mark directions with FIB



FIB sample preparation for EBSD and TEM

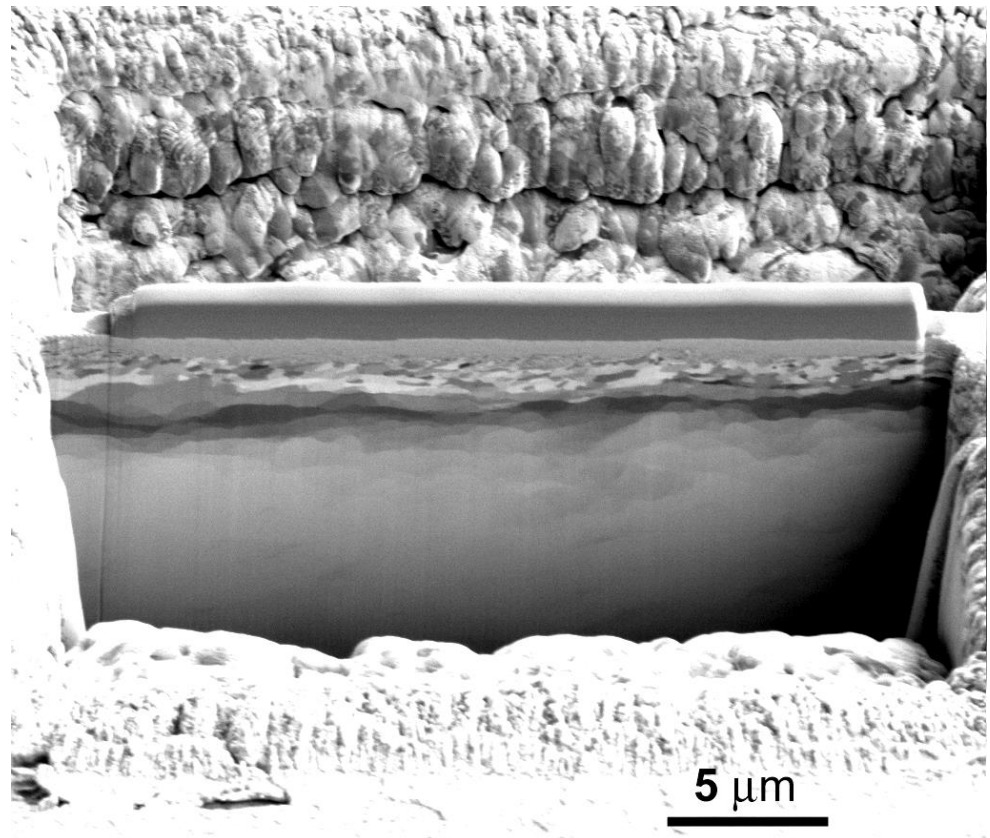


Wear scar

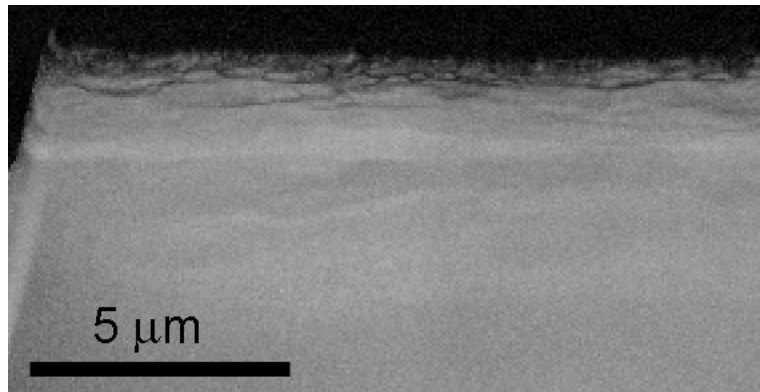
FIB sample

**Ion channeling
contrast image of
wear scar**

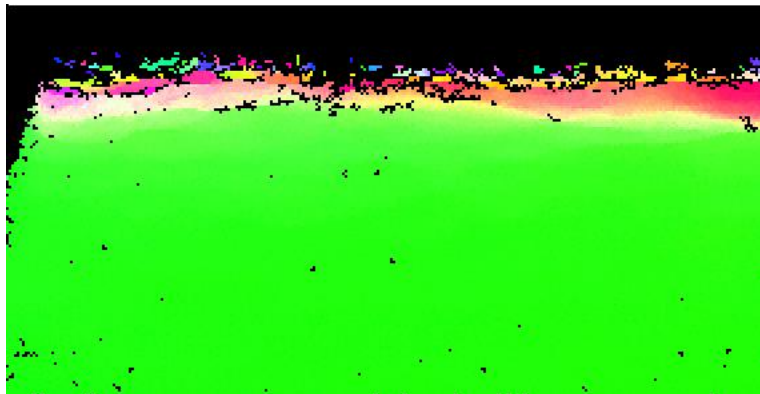
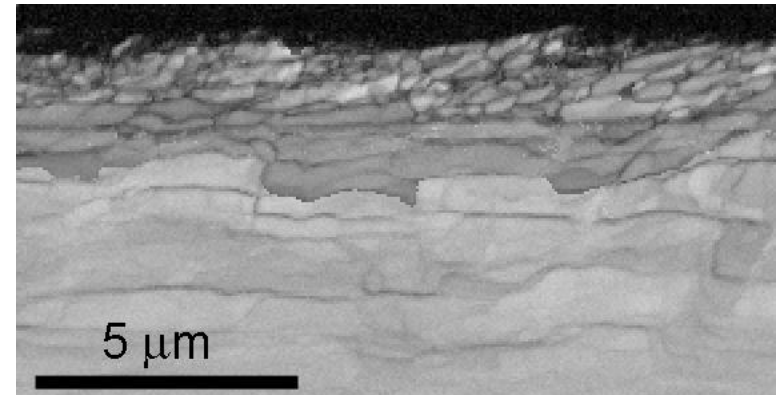
**<211> on (111) Ni
single crystal**



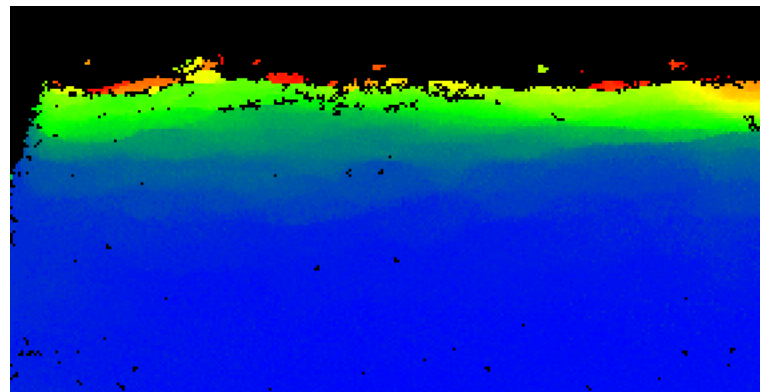
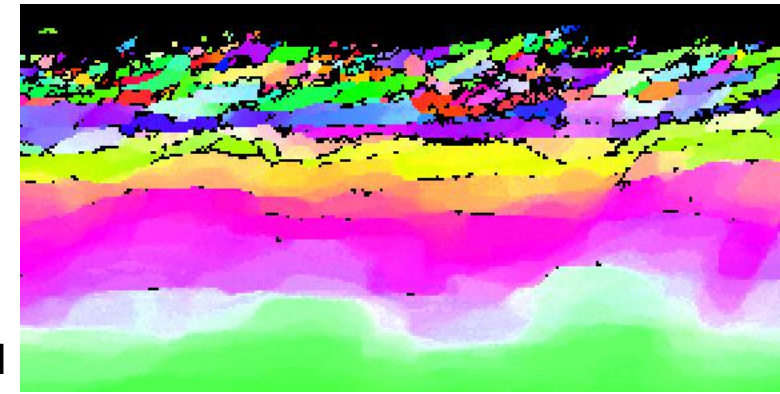
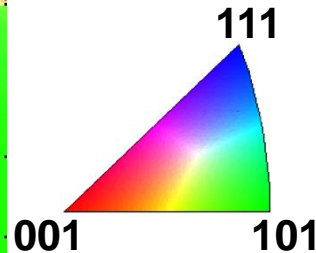
EBSD provides quantitative information ($\langle 110 \rangle$ on (111) Ni



Band
contrast



Sliding
direction

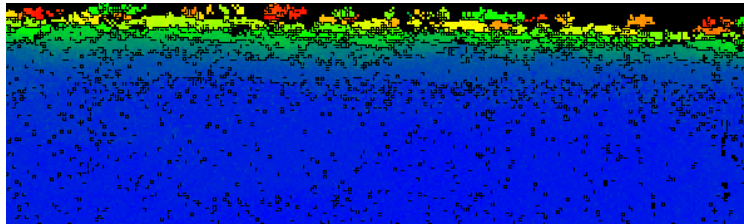


Orientation
difference

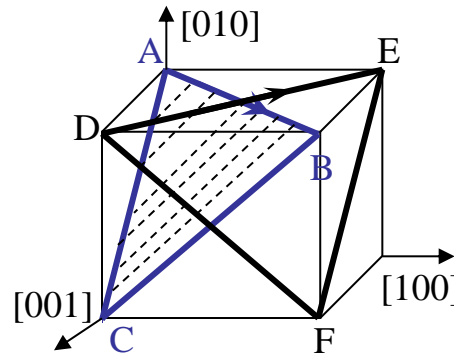
10 gram load for 1000 cycles

100 gram load for 1000 cycles

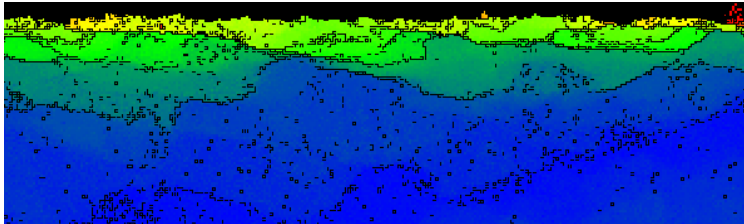
Relationship between crystallography, deformation and friction



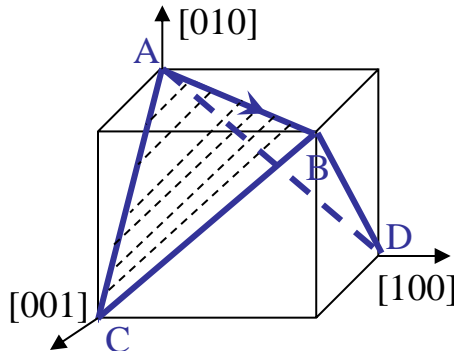
$\langle 100 \rangle (001)$



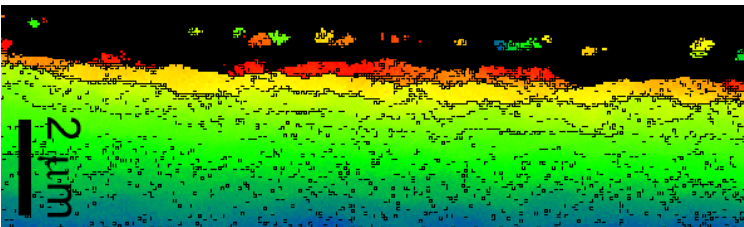
Strong dislocation interactions- high work hardening – low friction



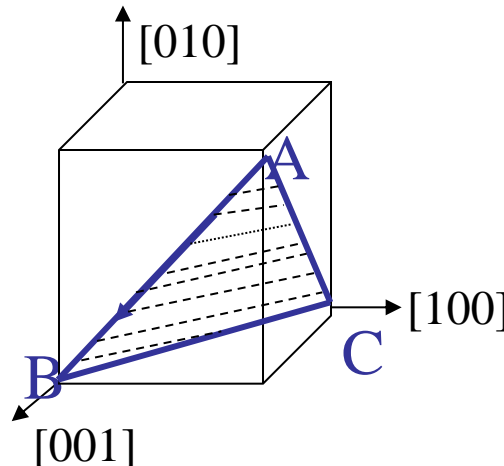
$\langle 110 \rangle (001)$



Weak dislocation interactions- low work hardening – high friction



$\langle 211 \rangle (011)$

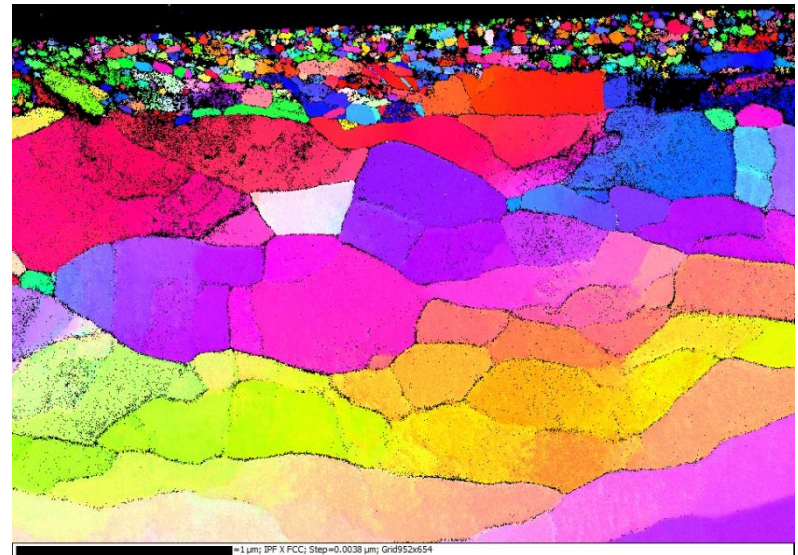
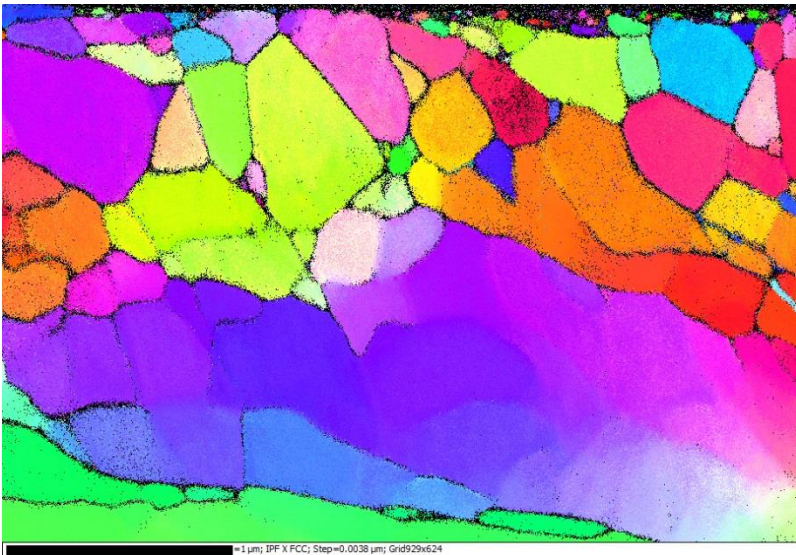
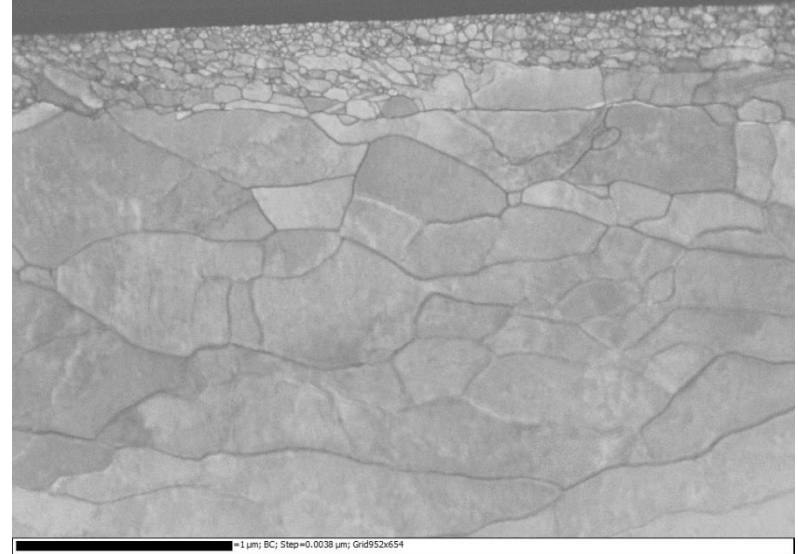
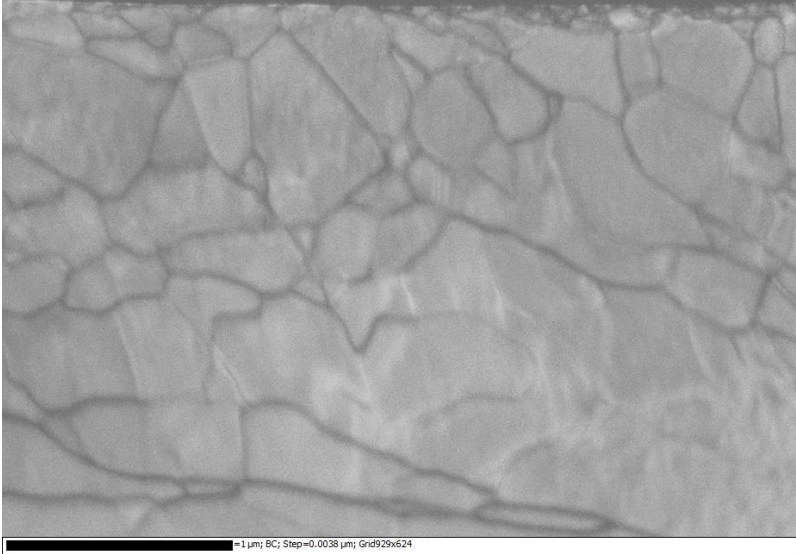


Very weak dislocation interactions- low work hardening – rapid recrystallization – low friction

Application of wear studies to electrical contacts (TKD)

1 gm load 1000 cycles

50 gm load 1000 cycles



FIB prepared thin samples TKD 4 nm step size

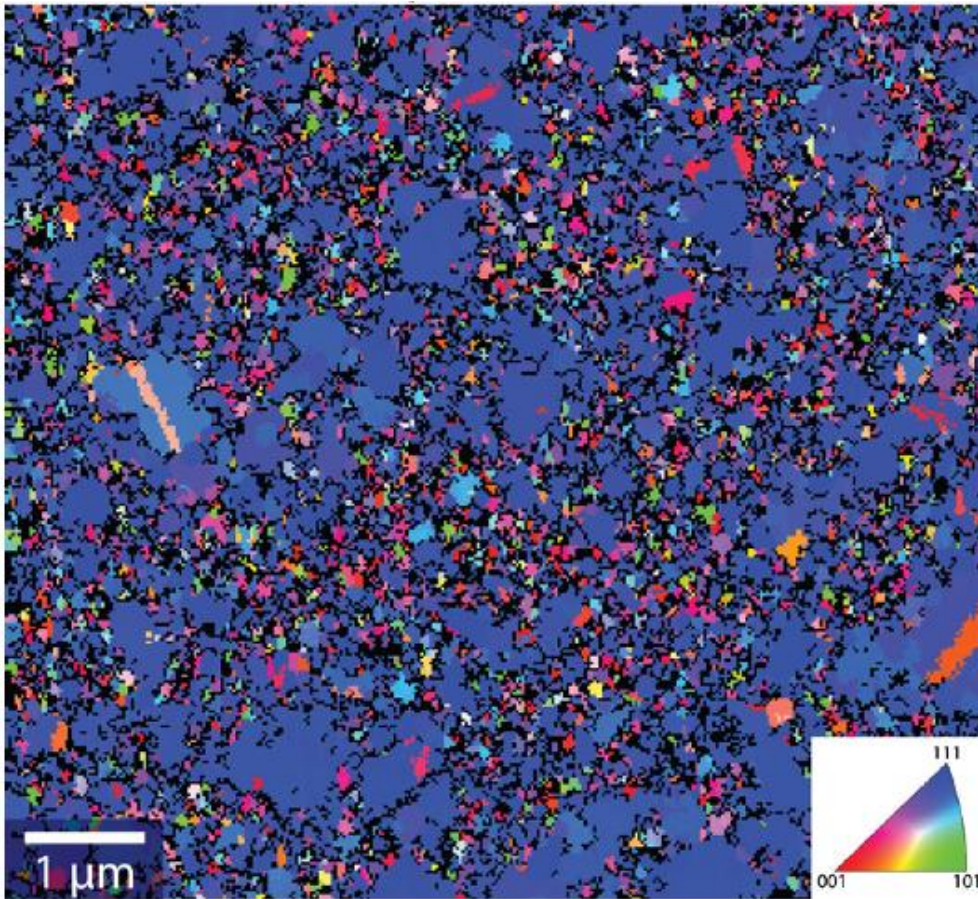
Development of PVD ZnO hardened Au for electrical contacts

- Inexpensive electrically conductive structural alloys are prone to oxidation and poor tribological performance.
- Electroplated hard gold thin films have been traditionally used to improve the tribological and contact resistance of these alloys.
- PVD provides an economical environmentally favorable approach to replace electroplated coatings
- Coatings must remain microstructurally stable during elevated temperature use
- Additions of ZnO through electron beam co-deposition provides a stable coating for long term applications in electrical contacts

For details see: Argibay, et al. *J. Appl. Phys.*, vol. 117, 2015, 145302.

Development of PVD ZnO hardened Au for electrical contacts

Typical pure Au microstructure obtained through e-beam deposition



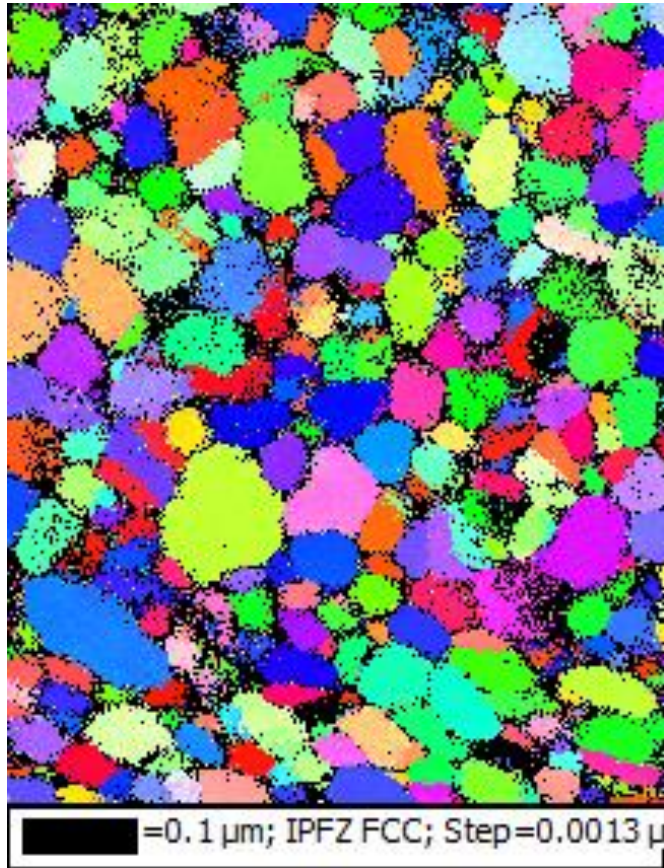
Highly textured coarse grained coatings that are prone to rapid grain growth are a result of pure Au deposition.

Microstructural instability can result in varying contact performance during long term use.

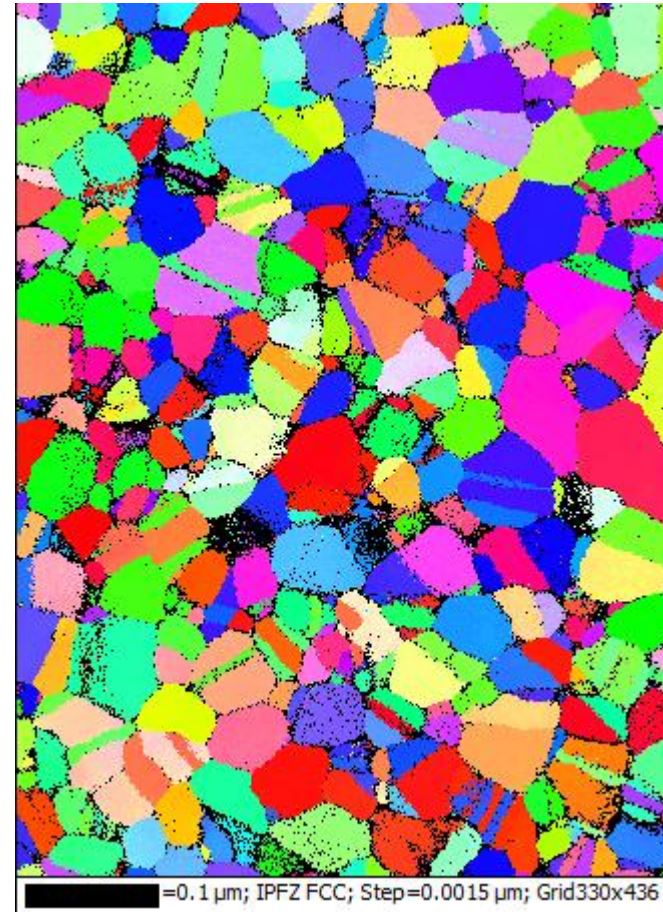
Development of PVD ZnO hardened Au for electrical contacts

As-deposited Au-ZnO microstructures

1 vol% ZnO



2 vol% ZnO

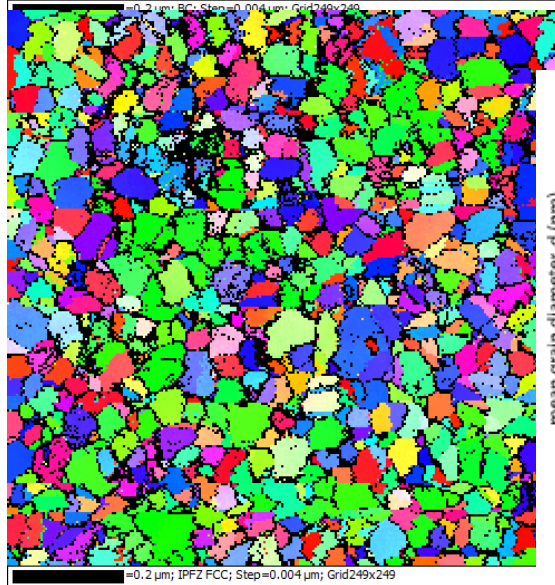
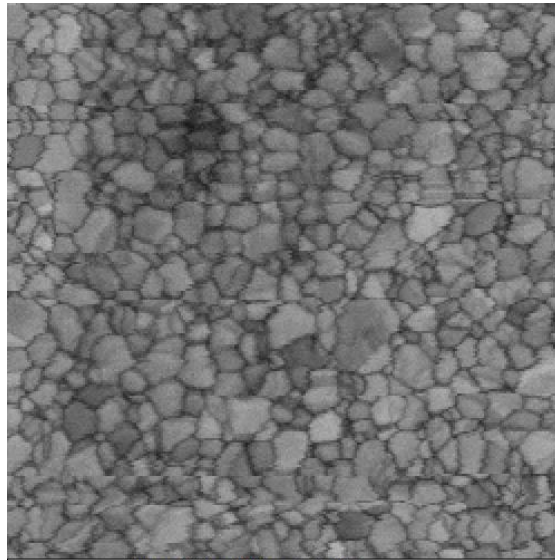


TKD was required to allow the grain structure to be observed. Note lack of texture in the as-deposited condition.

Plan view samples were prepared via Ga ion beam milling

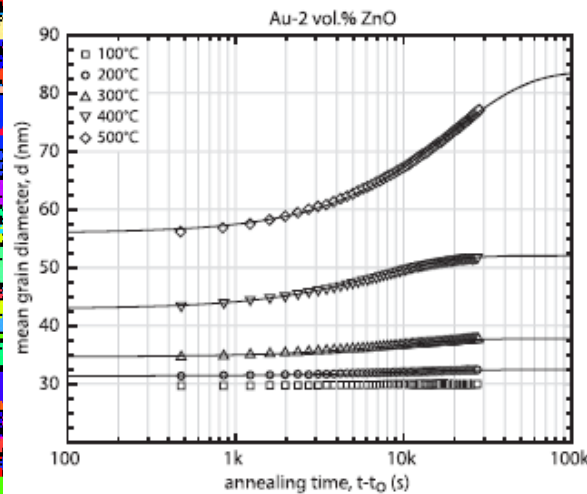
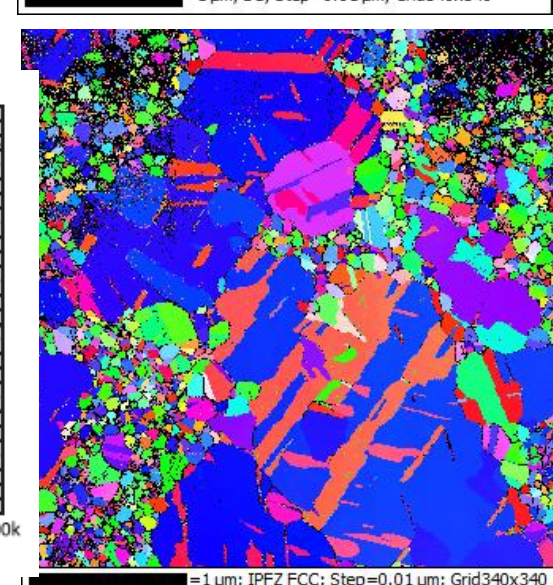
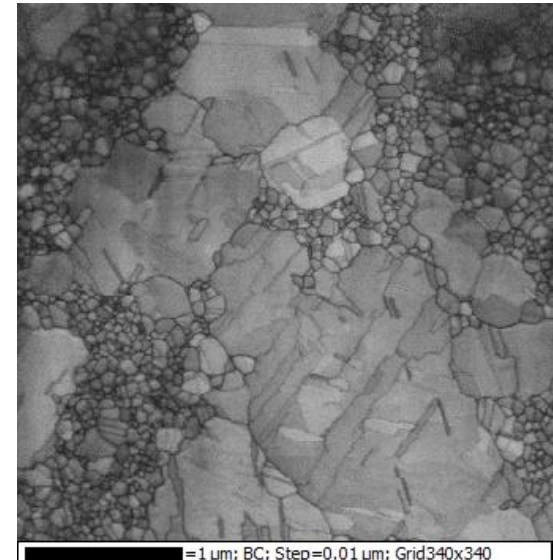
Development of PVD ZnO hardened Au for electrical contacts

200 C



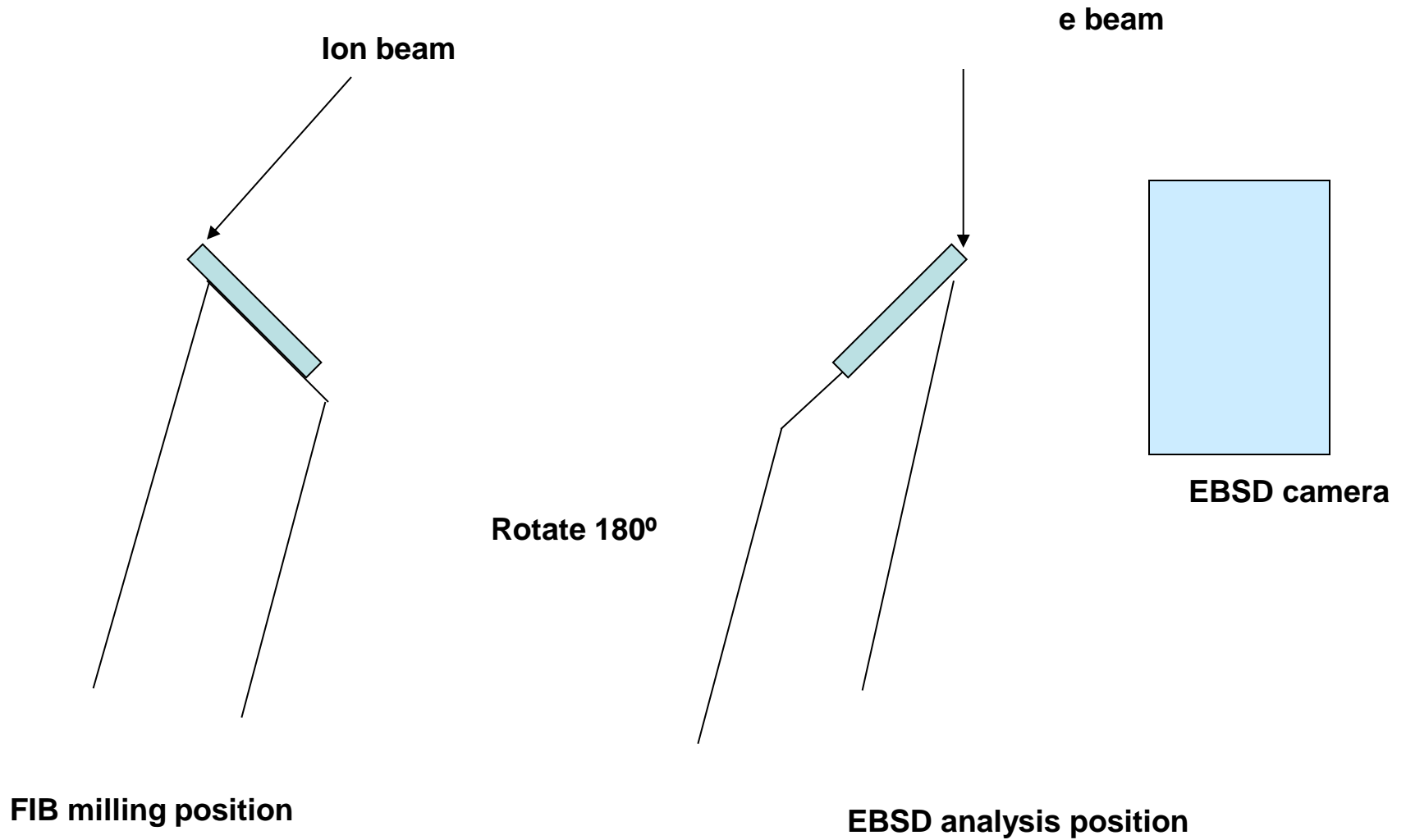
Au-2% ZnO exhibits slower grain growth kinetics as compared to typical hard gold coatings.

500 C



Plan view TKD of 2 % ZnO samples

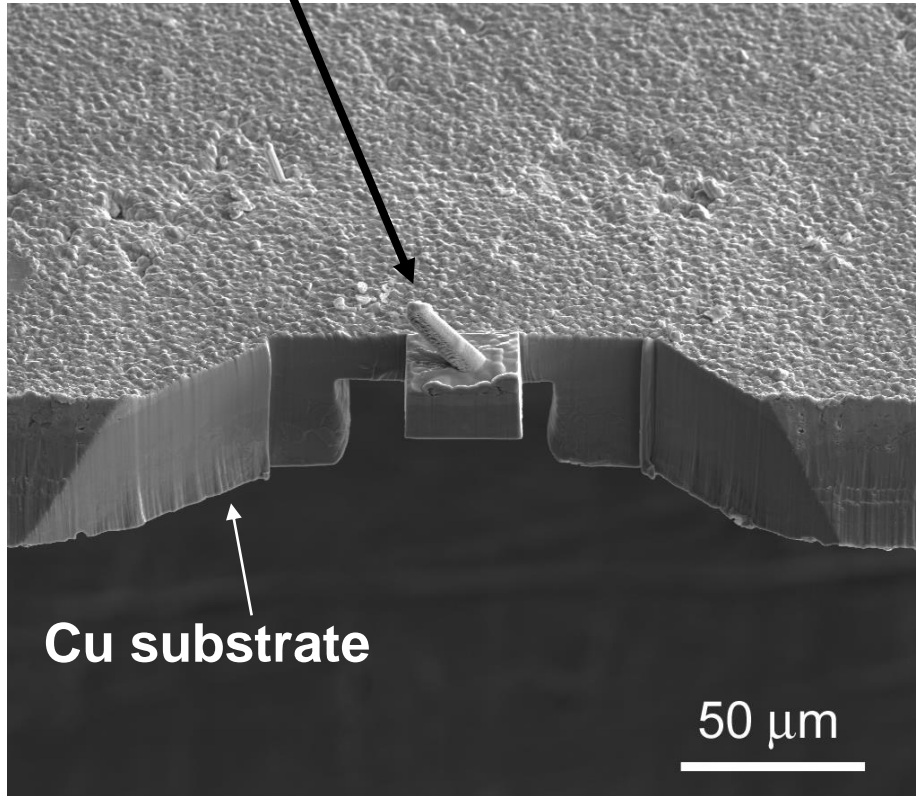
3D Electron Backscatter Diffraction



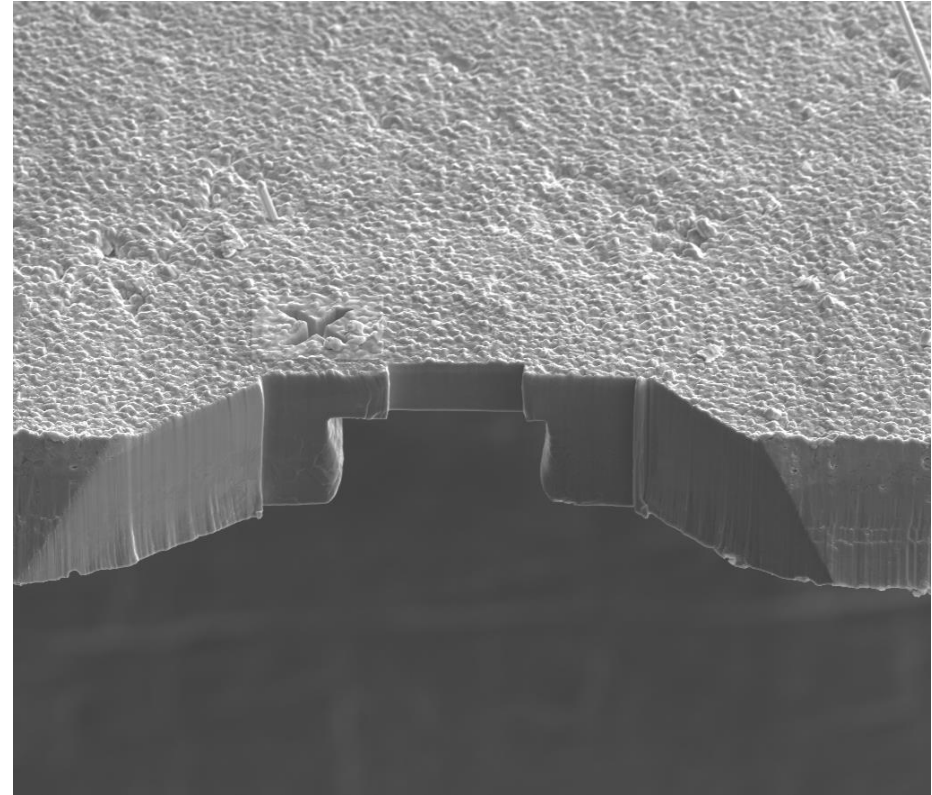
Geometry allows moving from FIB milling position to EBSD position with a simple 180° rotation.

FIB enabled 3D Electron Backscatter Diffraction

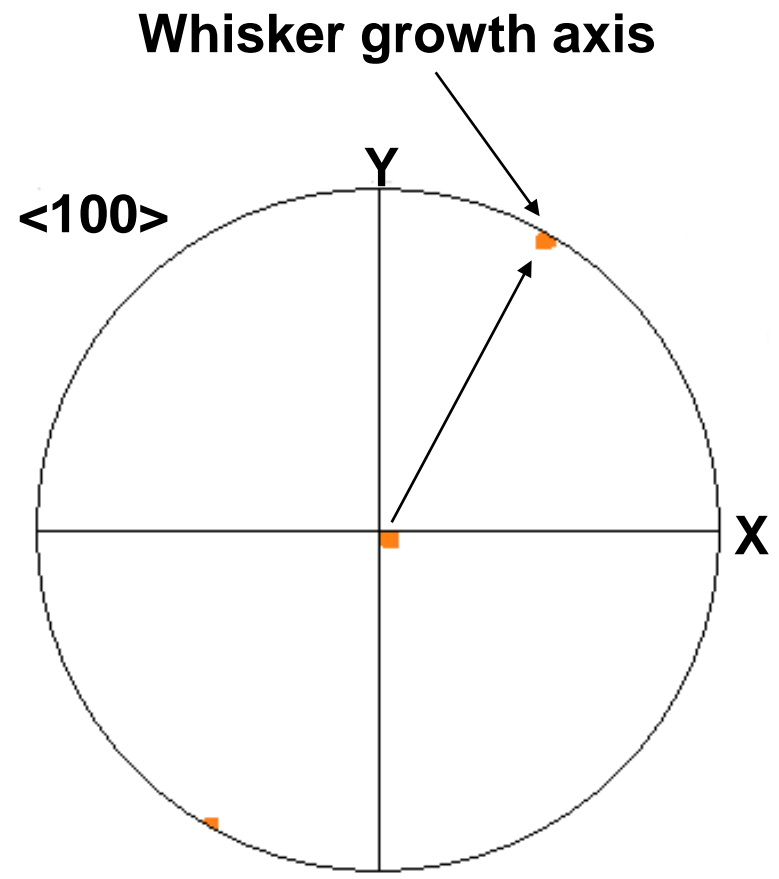
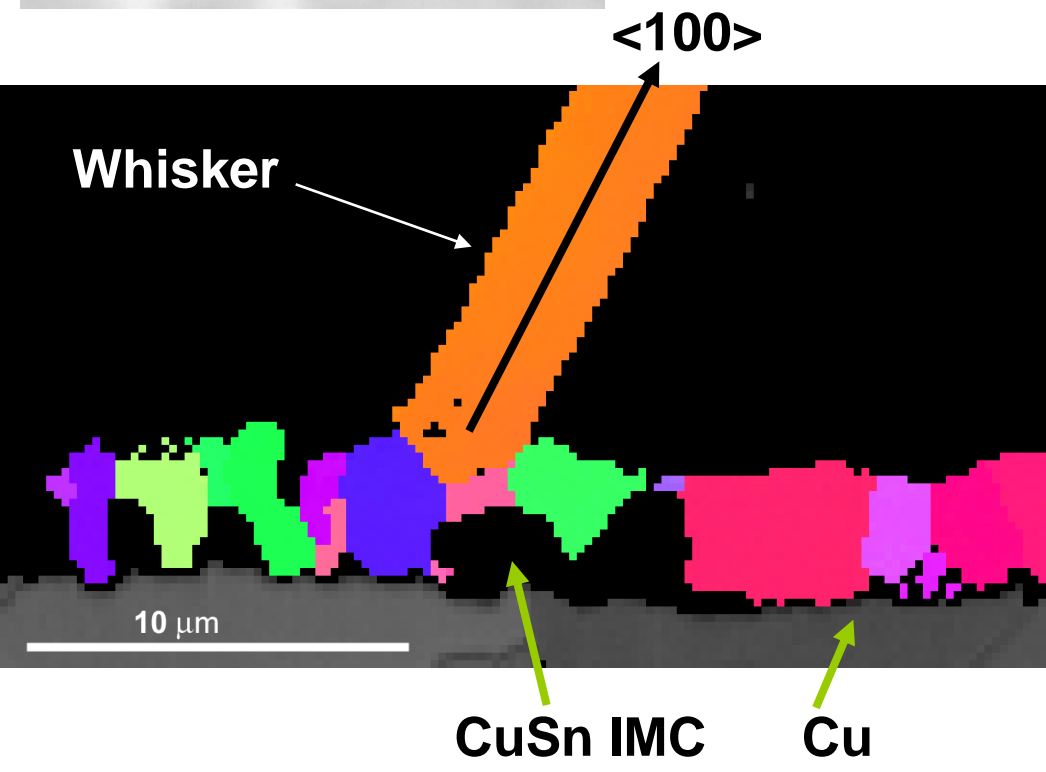
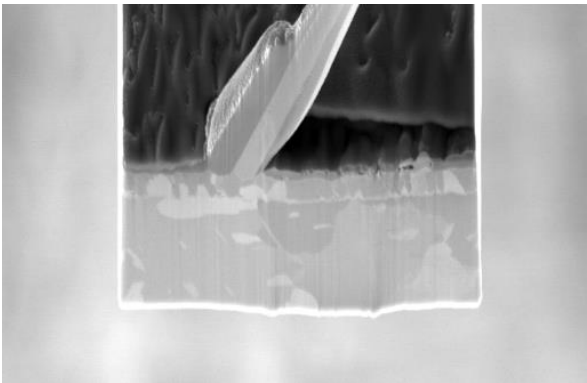
Whisker with Pt overlayer



Before 3D run

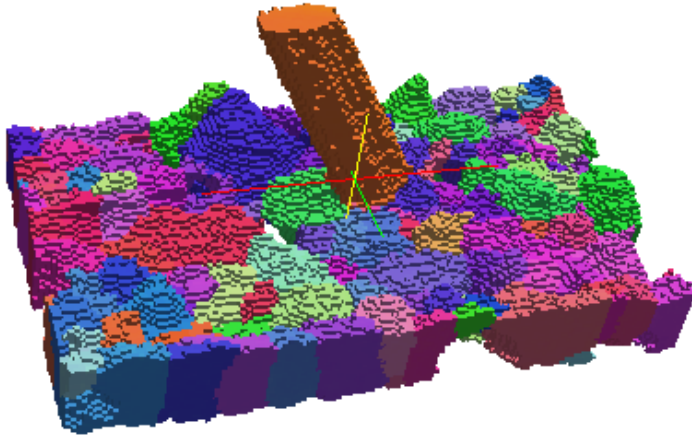


After 3D run

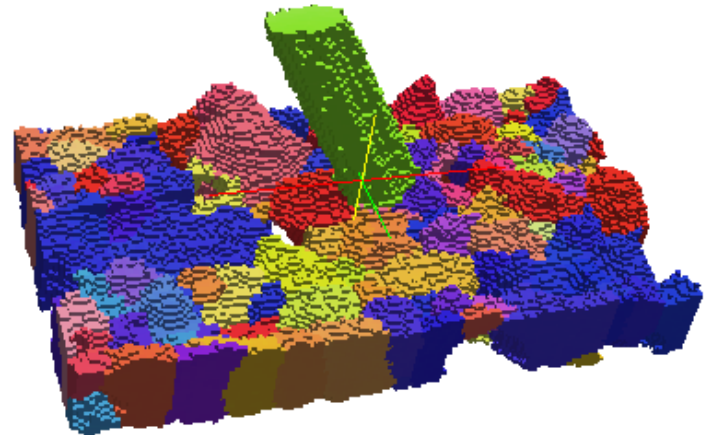


3D EBSD using FIB/SEM may help!

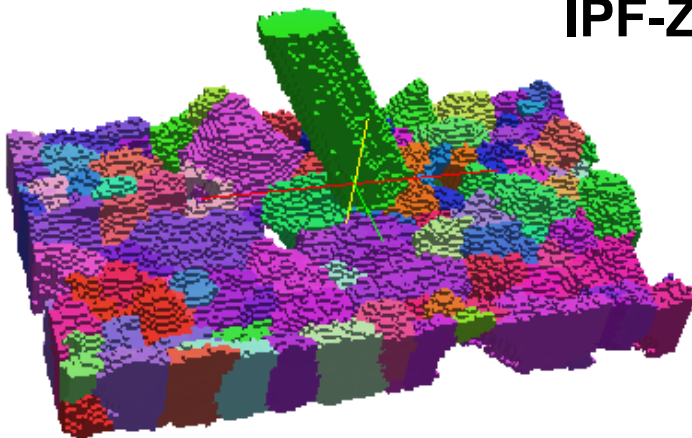
IPF-X



IPF-Y



IPF-Z

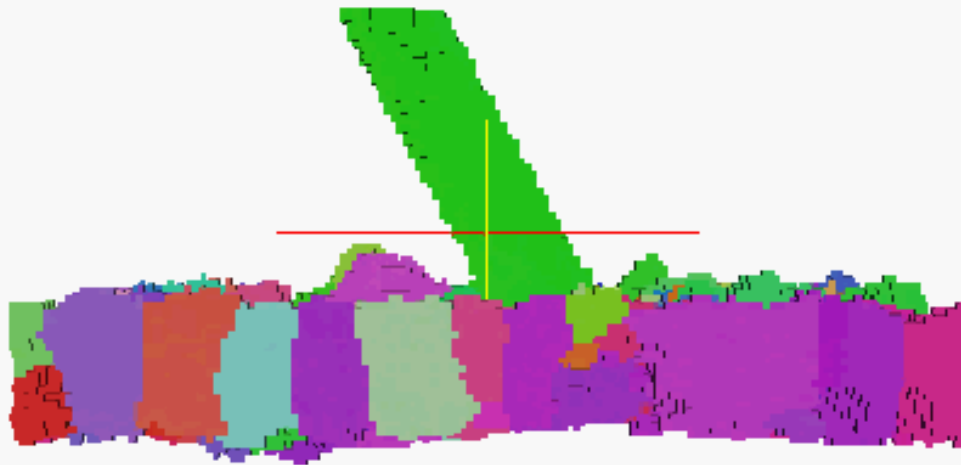


Data reconstructed from 75 slices 200 nm thick (each pixel 200 nm³).

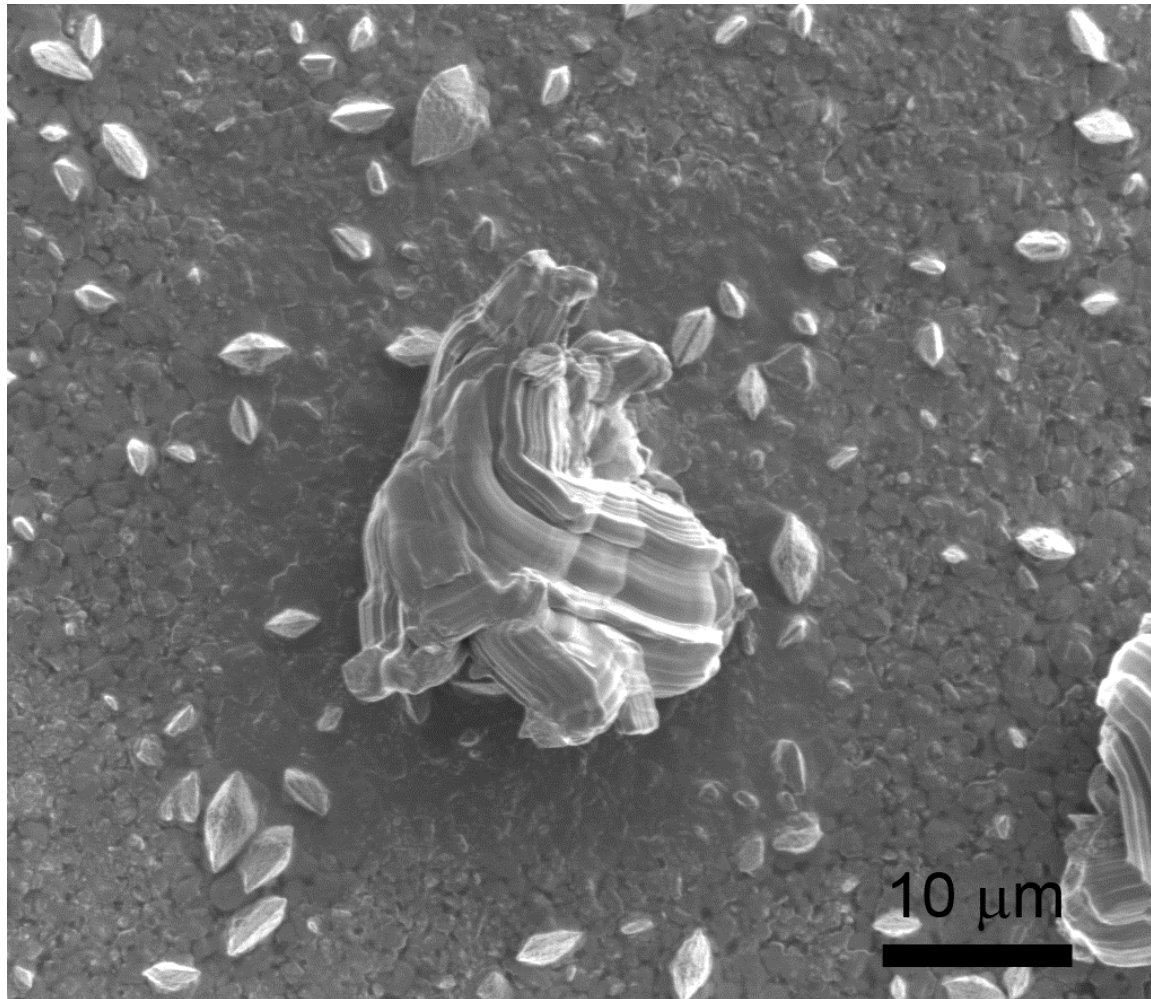
Total time was about 48 hours.

Data constructed using DREAM.3D

3D EBSD using FIB/SEM may help!

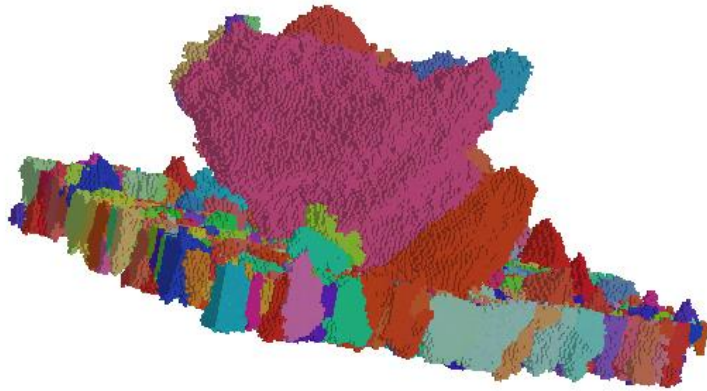


3D EBSD of large hillock on electroplated tin

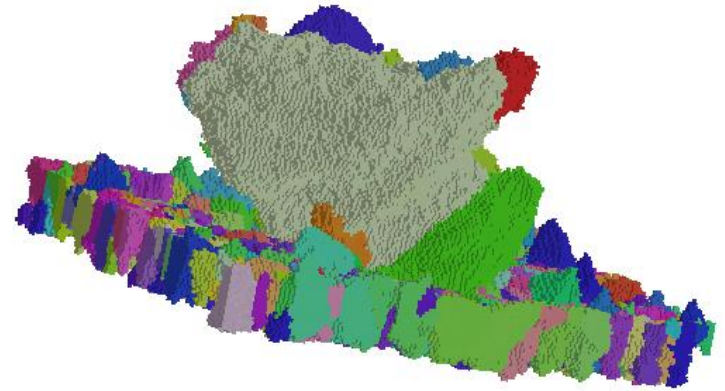


3D EBSD of large hillock on electroplated tin

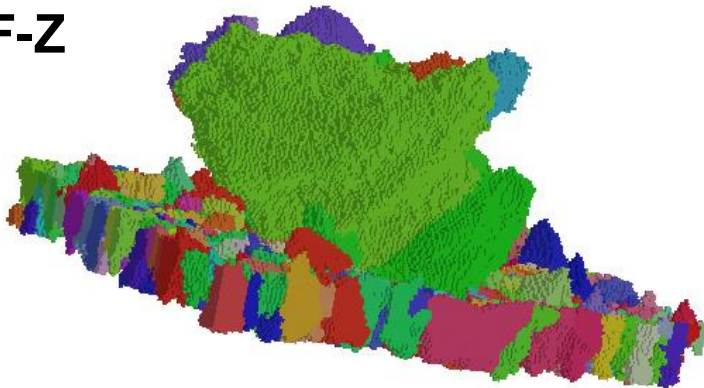
IPF-X



IPF-Y



IPF-Z

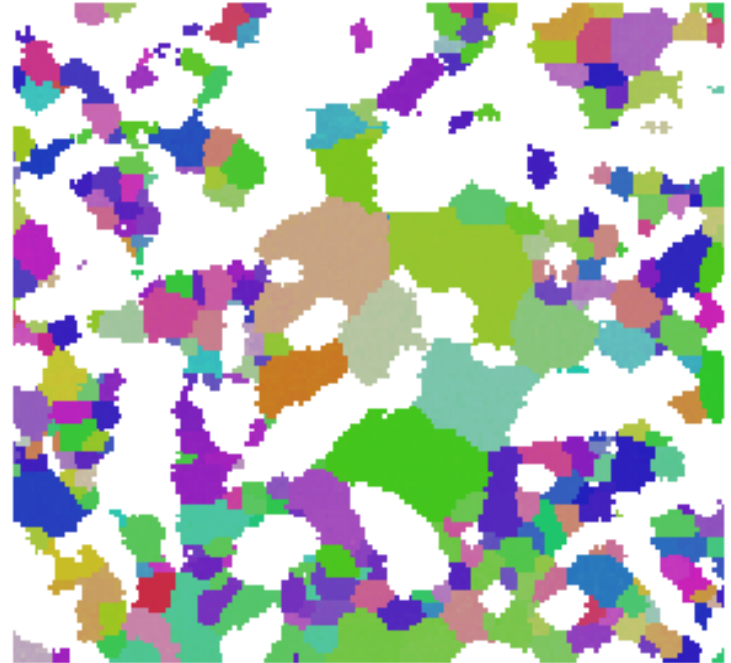


3D EBSD of large hillock on electroplated tin

Cross section



Plan view of tin film



Slices through 3D data of hillock demonstrates that the hillock is mostly a single crystal and there are large grains due to recrystallization in the tin film.

Application of EBSD in materials science:

Applies to many materials

- ✓ **Metals**
- ✓ **Ceramics**
- ✓ **Whiskers**
- ✓ **Mechanical testing**
- ✓ **Materials processing**
- ✓ **Use FIB for 3D EBSD**

Acknowledgements:

Meteorite studies:

Joe Goldstein

AM and Stainless Steel Artifacts:

Jeff Rodelas

Michael Maguire

Wear studies and Contact materials:

Somuri Prasad

Nic Argibay

Tin whiskers:

Don Susan

Expert technical help:

Bonnie McKenzie –SEM and EBSD

Alice Kilgo - Metallography

Michael Rye – FIB sample preparation