

Mitigating Structural Defects in Droop-Minimizing InGaN/GaN Quantum Well Heterostructures

Zhibo Zhao¹, Jordan Chesin¹, Akshay Singh¹, Erik Nelson², Isaac Wildeson², Parijat Deb², Andrew Armstrong³, Eric Stach⁴,
Silvija Gradečak¹

¹ Massachusetts Institute of Technology, Cambridge, MA

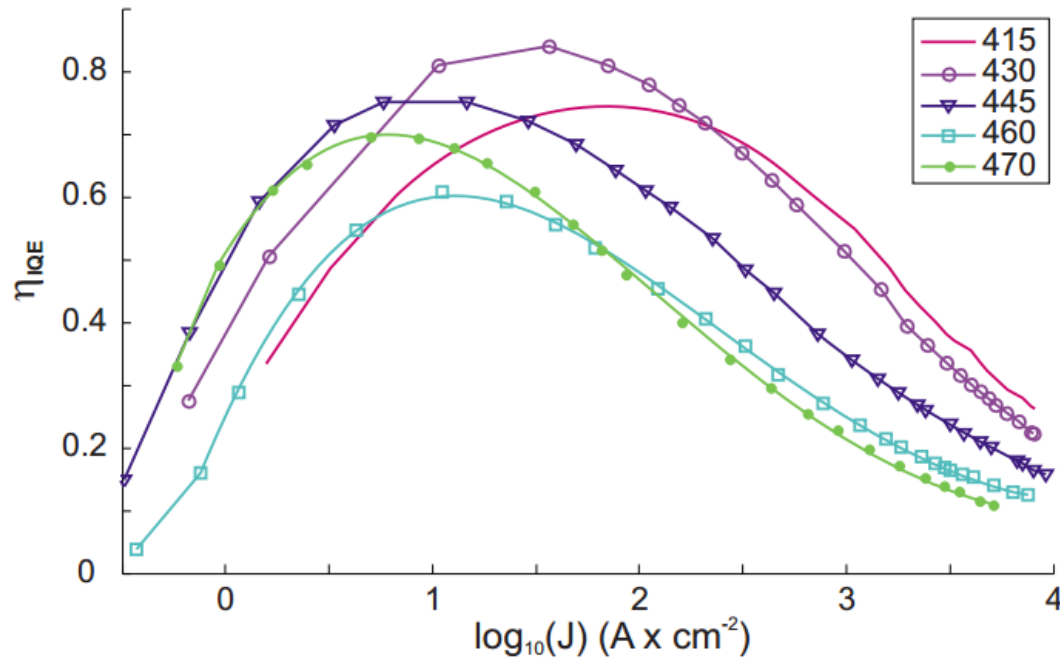
² Lumileds Lighting, San Jose, CA

³ Sandia National Laboratories, Albuquerque, NM

⁴ Brookhaven National Laboratories, Upton, NY

Efficiency Droop in III-Nitride LEDs

Wavelength-dependent droop measurements in InGaN QWs

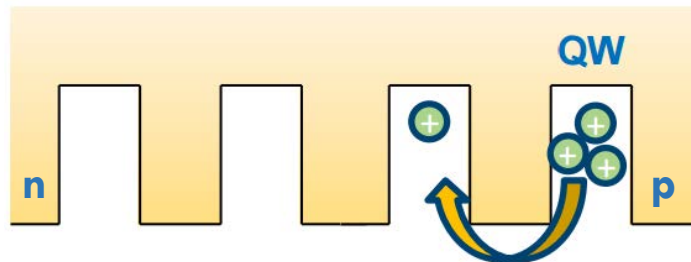


III-Nitride LEDs continue to suffer from efficiency losses at high current:

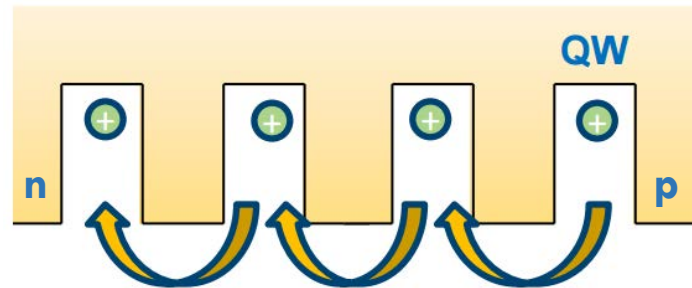
- Mechanism likely arises from Auger recombination
- Significant gains in flux per dollar/watt remain to be realized
- Droop is more significant at longer wavelengths

Strategies to Address Efficiency Droop

1) Improve carrier spreading across quantum wells



Standard active region design:
carriers localized to p-side wells



Improved hole transport design:
carriers distributed more evenly

2) Reduce QW thickness to enhance radiative recombination rate

$$R = An + Bn^2 + Cn^3$$

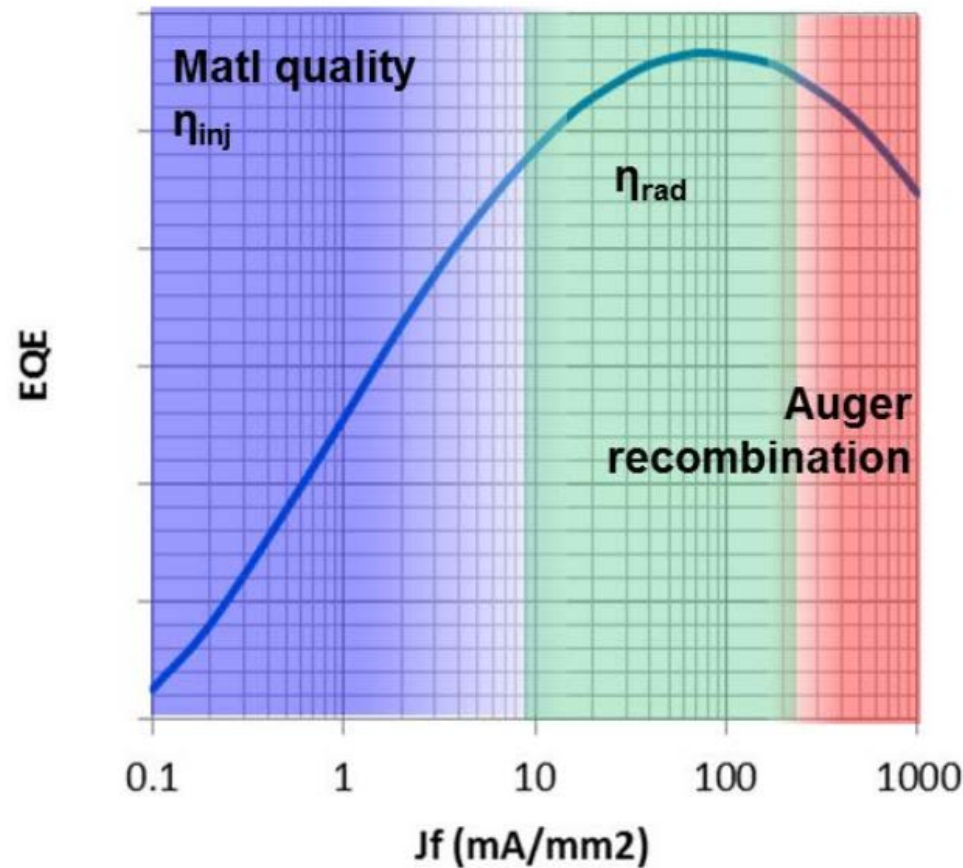
Increase in **B** reduces steady-state carrier concentration

3) Increase QW thickness to reduce Auger matrix element

$$M = \iint d^3\vec{r}_1 d^3\vec{r}_2 \phi_1^*(\vec{r}_1) \phi_2^*(\vec{r}_2) \frac{\exp[-\lambda r_{12}]}{r_{12}} \phi_1(\vec{r}_1) \phi_2(\vec{r}_2)$$

Thicker QWs → More screening → Screening length (λ^{-1}) shorter → Smaller M

Understanding Impacts of Material Quality

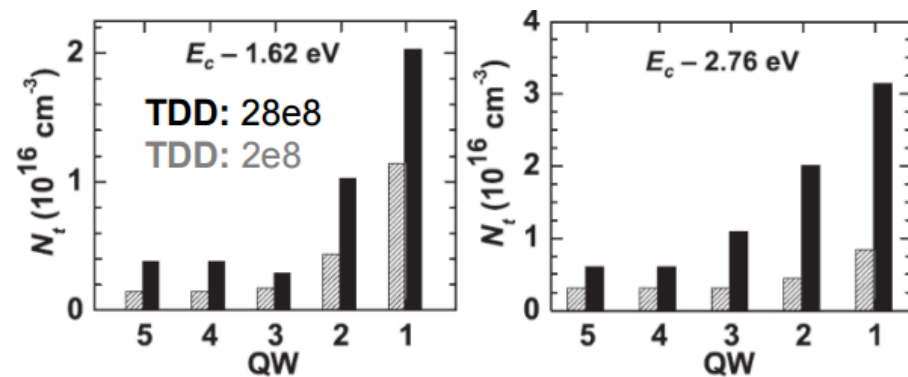


Different droop mitigation strategies impact different regions of the efficiency curve BUT interlinked physical processes in the QWs make it difficult to **improve droop** without hurting **material quality**

Understanding Impacts of Material Quality

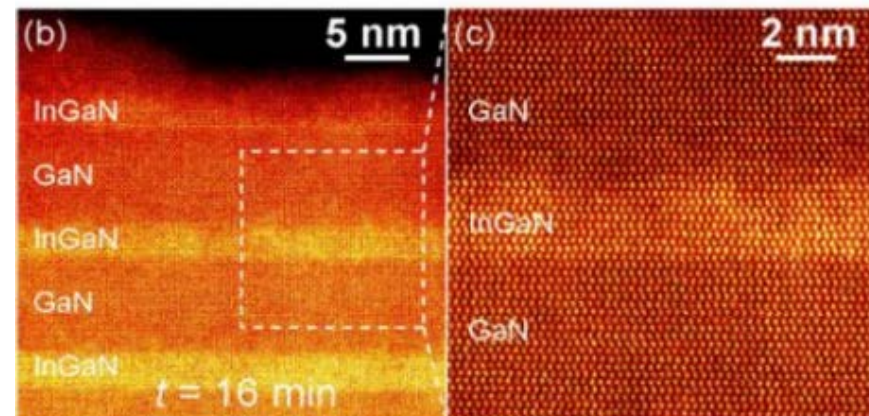
Need for intelligent design process informed by fundamental material characterization:

DLOS: electronic nature of defects



- Energy and density of defects present in heterostructure
- Bias/depth-dependent DLOS provides limited spatial resolution

STEM: atomistic nature of defects



- Atomically resolved imaging and compositional analysis
- Artifact-free under low-dose imaging conditions

S/TEM Imaging Parameters

A. Aberration-corrected FEI Titan S/TEM

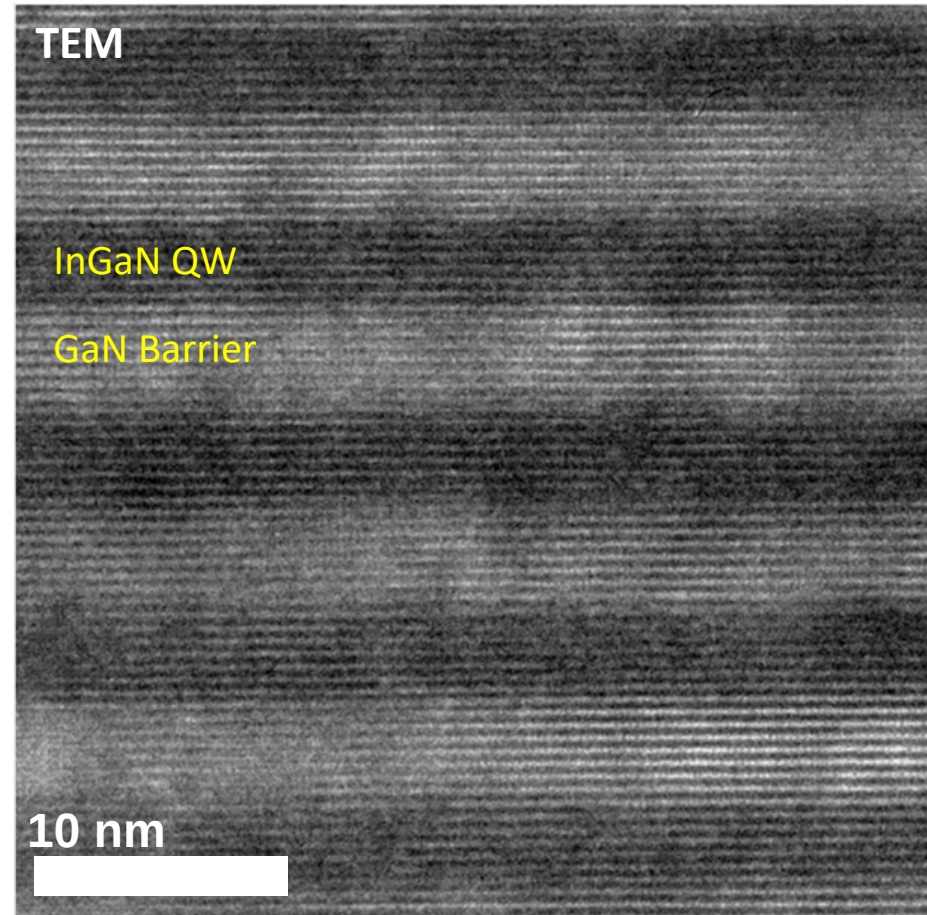
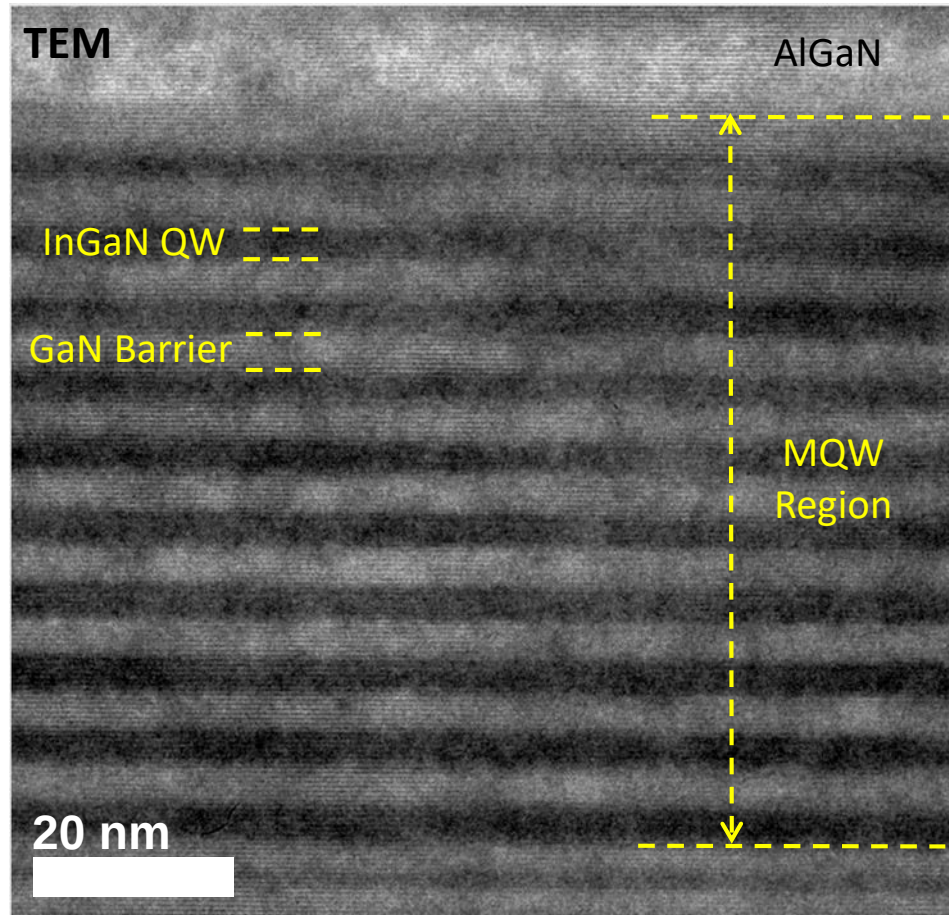
- Sample preparation: FIB milling (2 kV)
- Accelerating Voltage: 80 kV
- Imaging modes: bright-field TEM and HAADF STEM
- Enables acquisition of artifact-free TEM micrographs below knock-on threshold for beam damage

B. JEOL 2010F Field Emission Gun S/TEM

- Sample preparation: FIB milling (2 kV)
- Accelerating Voltage: 200 kV
- Imaging modes: bright-field TEM and HAADF STEM

Growth A: SRH Penalty

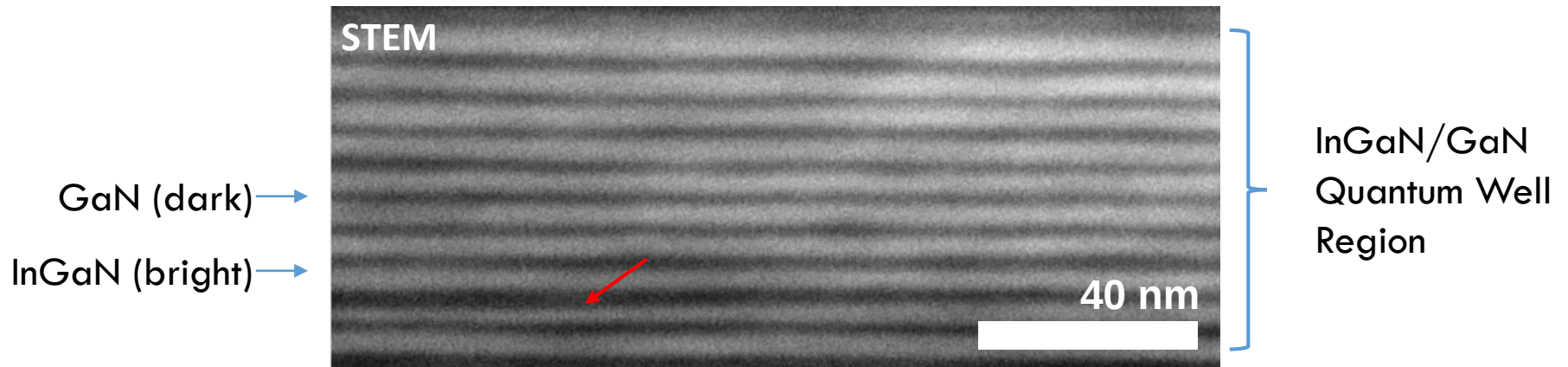
Structural changes result in a measured performance penalty in low current regimes



No obvious structural defects were observed in TEM imaging

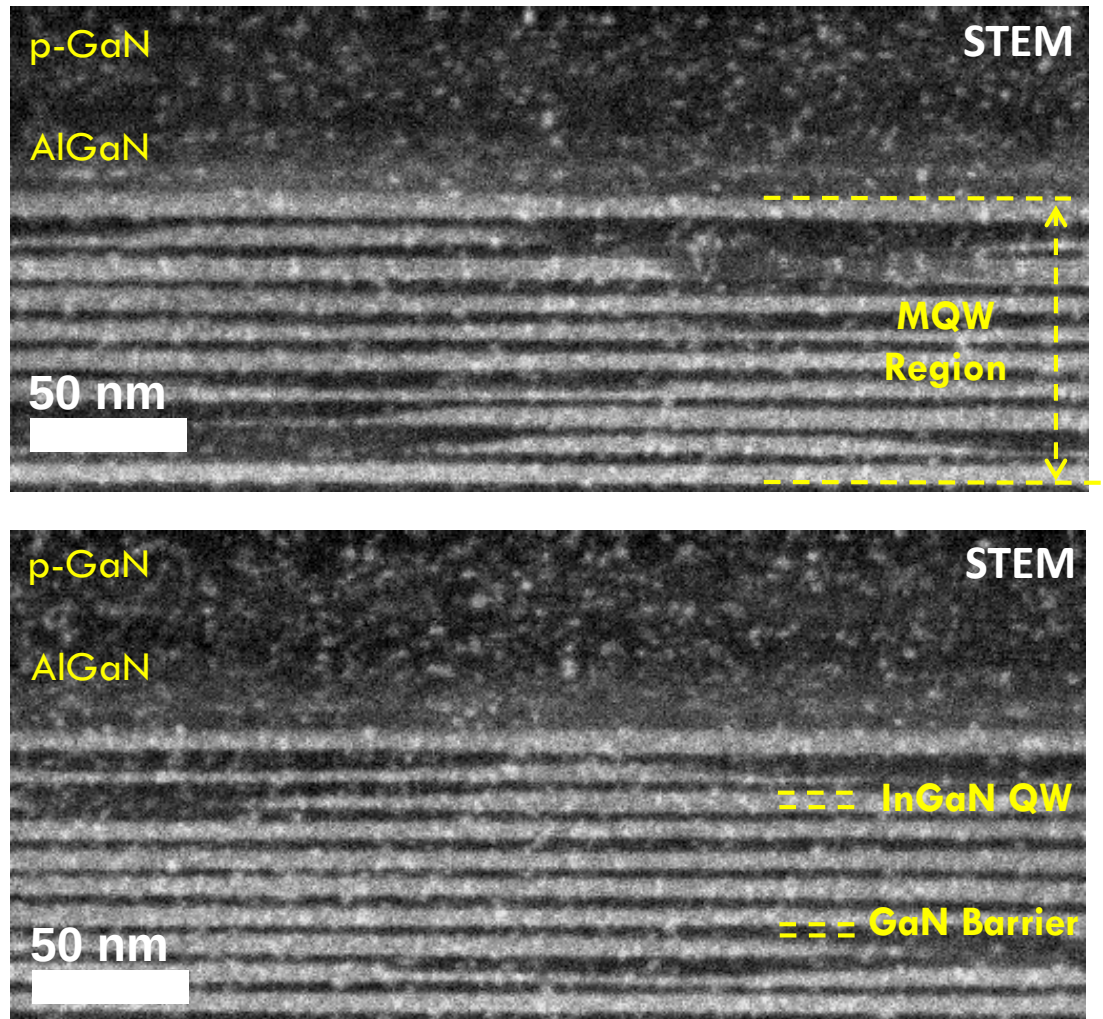
Growth B: Severe SRH Penalty

Growth process modified to exaggerate detrimental structural changes



Observed gross well width fluctuations (WWFs) in HAADF STEM

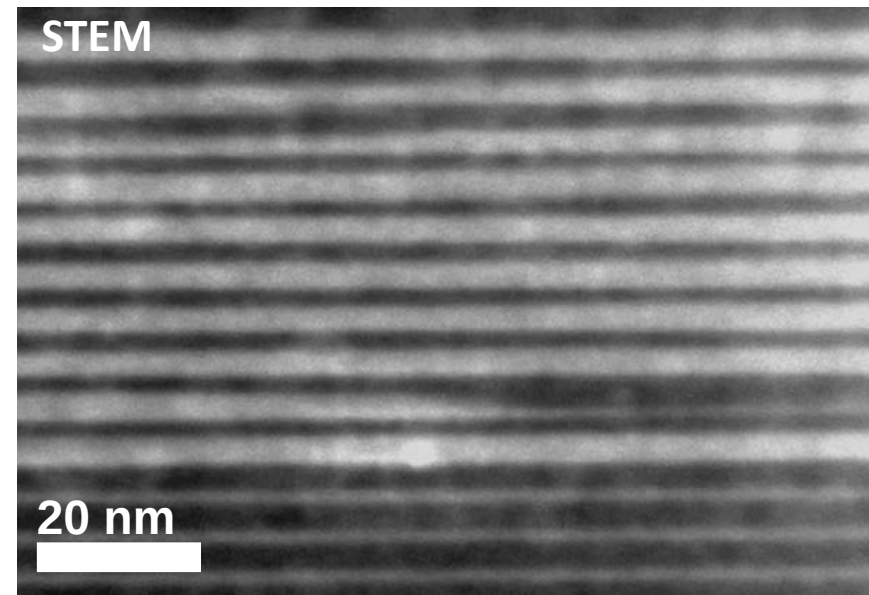
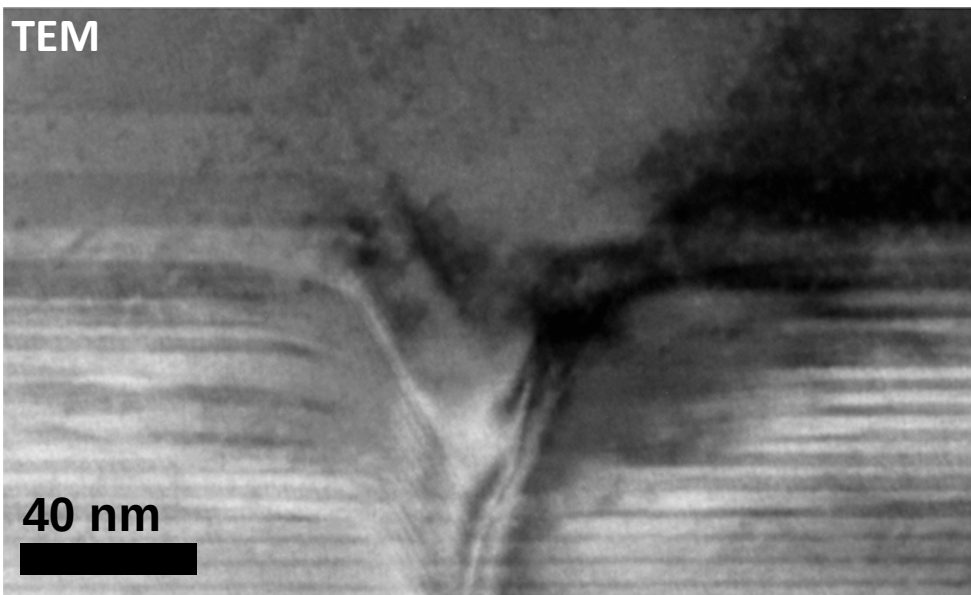
Growth B: Severe SRH Penalty



HAADF STEM imaging performed at 80 kV rules out the possibility that the observed WWFs are artifacts of beam damage

Growth B: Severe SRH Penalty

Presence of gross WWFs were observed concurrently with V-pit defects opening in the QW active region



Suggests growth process is sensitive to local variations in strain: local alloy composition may vary as a result of strain-based In incorporation

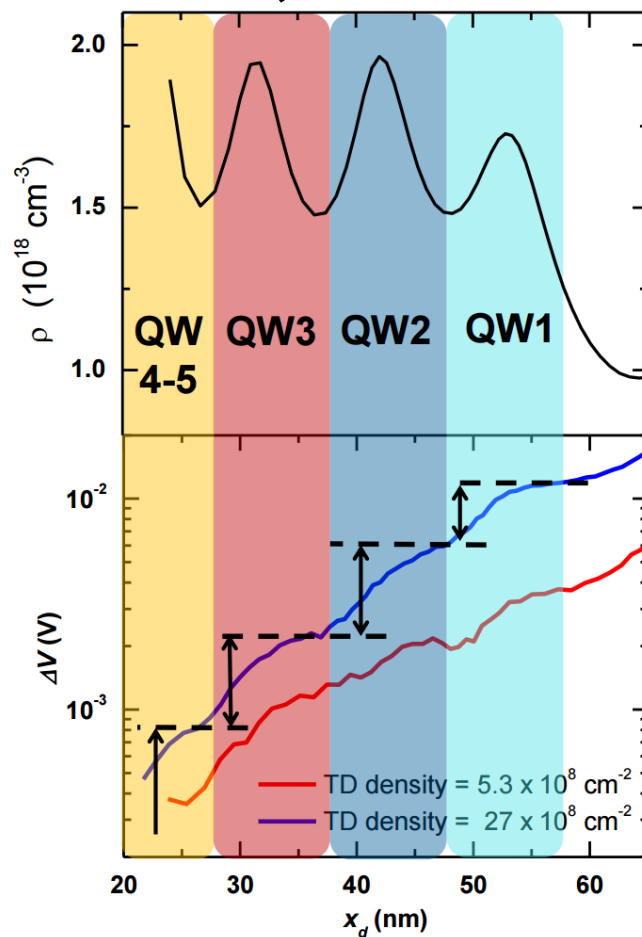
Defect Levels via Deep-level Optical Spectroscopy

Deep-level optical spectroscopy (DLOS)

- Can directly probe energy levels of defects present in heterostructure
- Defect density can be calculated from lighted C-V transients
- Bias/depth-dependent DLOS provides limited spatial resolution

DLOS shows high densities of a midgap defect state within the active region which correlates with measured IQE

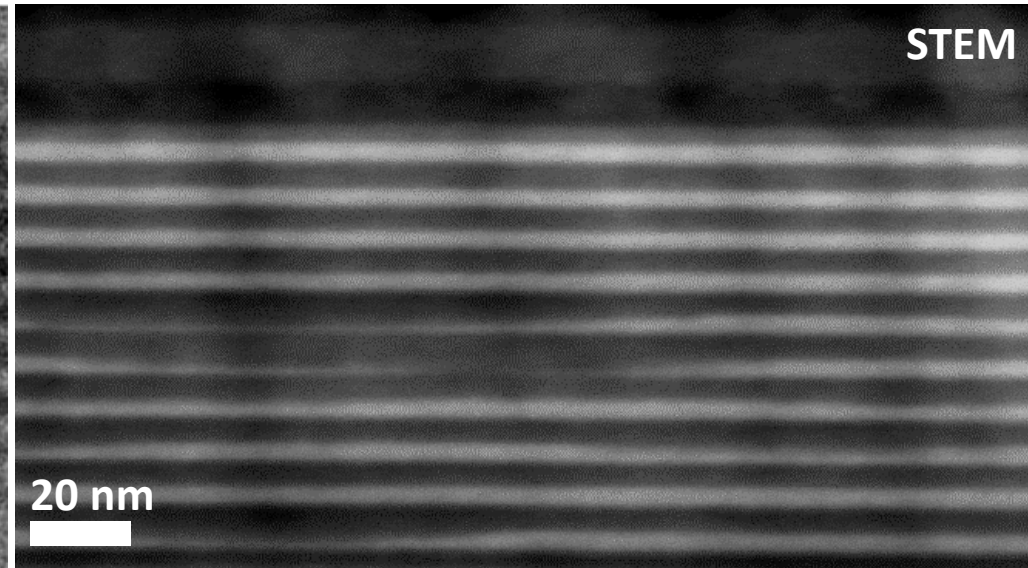
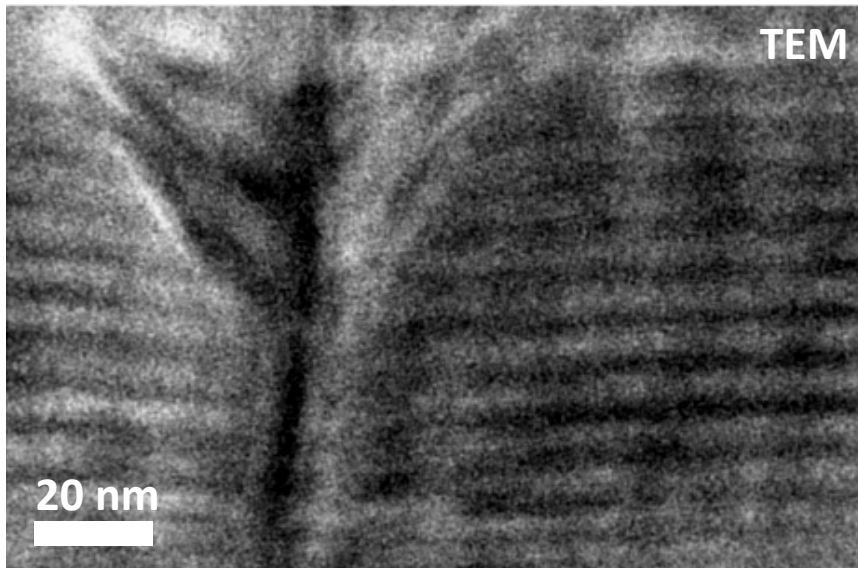
Example of Depth Profile of MQW Defects



Do the observed well-width fluctuations correlate with performance?

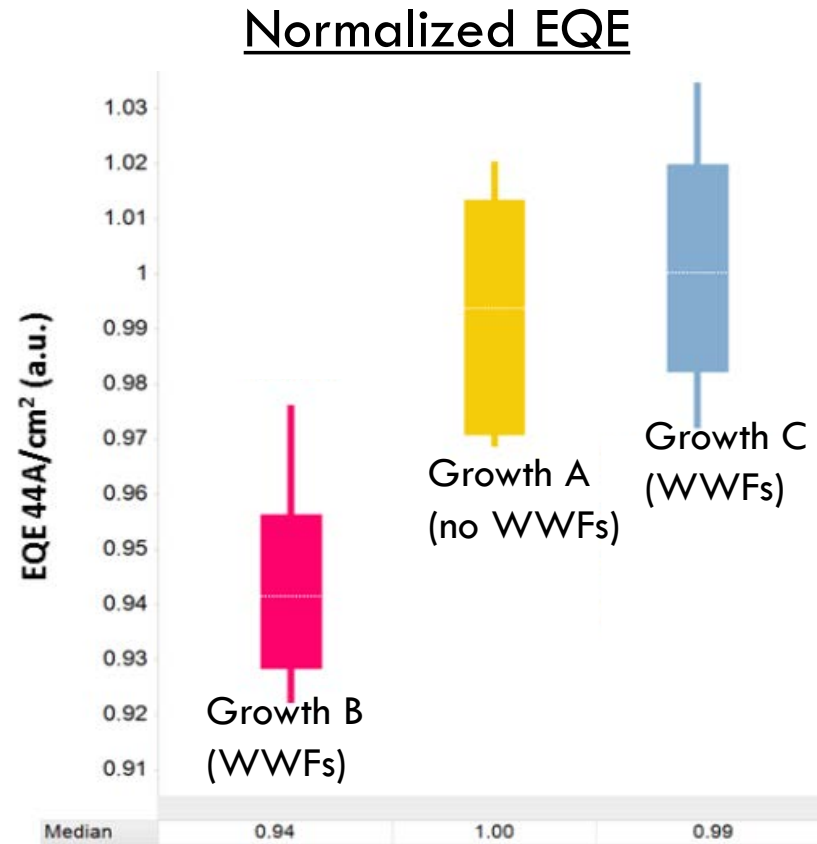
Growth C: Low SRH Penalty

Structure and measured performance similar to Growth A (no WWFs) but grown with process similar to Growth B (gross WWFs)



S/TEM images again demonstrate correlation between V-pits opening up in QW region and nearby WWFs

Performance Comparison Between Growths



EQE performance does not seem to be entirely dominated by WWFs

Conclusions and Future Work

Conclusions

- Observed QW width fluctuations arising from growth changes in droop-minimizing epitaxial InGaN/GaN heterostructures
 - Presence of WWFs correlated with V-pits opening in QW region
 - May be a result of strain-dependent In incorporation into QW layer
- QW width fluctuations do not dominate EQE performance
 - WWFs are not necessarily indicative of severely compromised material quality
- DLOS suggests dominant defects are localized to QW region

Future/Ongoing Work

- Current sample set: no WWFs but still exhibit performance penalty
 - C_s -S/TEM and EELS measurements at Brookhaven National Lab
 - Atom probe tomography: local composition and interface quality
 - Cathodoluminescence in STEM: directly correlate microstructural features with optical emission

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