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Experiments to characterize particle flotation in a curing epoxy

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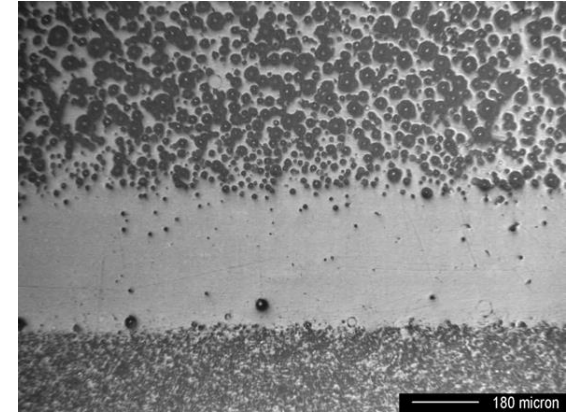
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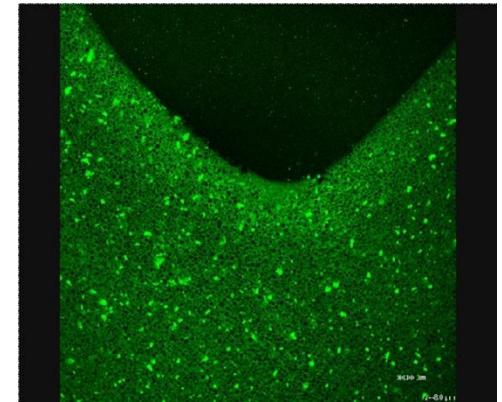
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Introduction

- Epoxies are used to encapsulate electronic parts to give structural integrity and protect from vibration
- Filled with glass microballoons to reduce thermal stresses
- End process, so high value
 - Encapsulation defects, e.g. voids, fracture, delamination and property inhomogeneities, can lead to component failure
- Complex materials: polymerizing, high fraction of solids of various sizes
- Non-isothermal: exothermal reactions and oven cure
- Rheological depend on temperature, extent of reaction, particle concentration, surface interactions, shear rate, shear history, time
- Particle concentration can change locally due to buoyancy or shear
- Predict distribution of filler particles as a result of processing time and temperature changes



An optical micrograph of a final product shows a clear layer next to a surface.

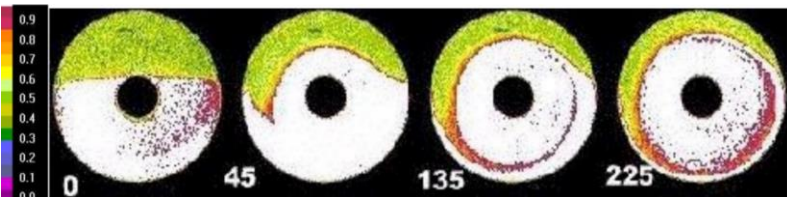
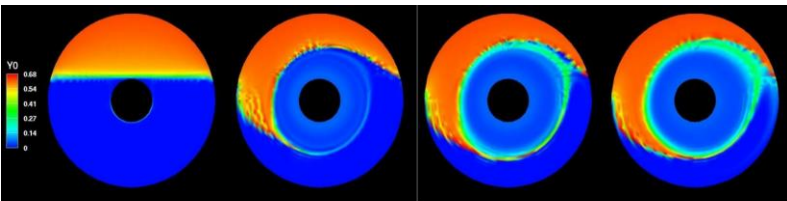
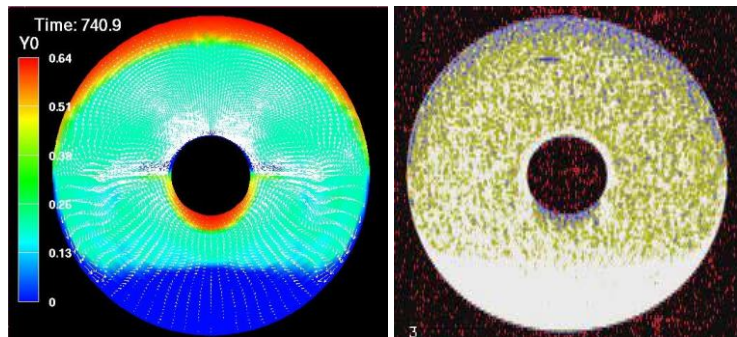


Confocal microscopy during wicking in an underfill of an integrated circuit shows small particles congregate near the advancing interface.

Outline

- Description of current engineering model of particle migration
- Particle Characterization and hindered settling
- Epoxy kinetics
- Rheology
- Example model results
- Needed extensions to the model and next steps

Diffusive Flux Model for Buoyant Particles in Curing Polymer



$$\rho \frac{\partial \mathbf{v}}{\partial t} + \rho \mathbf{v} \cdot \nabla \mathbf{v} + \nabla p - \nabla \cdot (\eta (\nabla \mathbf{v} + \nabla \mathbf{v}^T)) - (\rho_f - \rho_s) \phi \mathbf{g} = 0$$

$$\rho = (1 - \phi) \rho_f + \phi \rho_s$$

$$\frac{\partial \xi}{\partial t} = (k_1 + k_2 \xi^m) (1 - \xi)^n$$

$$\eta = \eta(\phi, \xi, T) = \eta_0(T_g) \left(1 - \frac{\phi}{\phi_{\max}}\right)^n \left(1 - \left(\frac{\xi}{\xi_c}\right)^m\right)^{-p} 10^{\frac{c_1(T - T_g)}{c_2 + T - T_g}}$$

$$\frac{\partial \phi}{\partial t} + \nabla \cdot (\phi \mathbf{v}) = \frac{\nabla \cdot \mathbf{J}_s}{\rho_s}$$

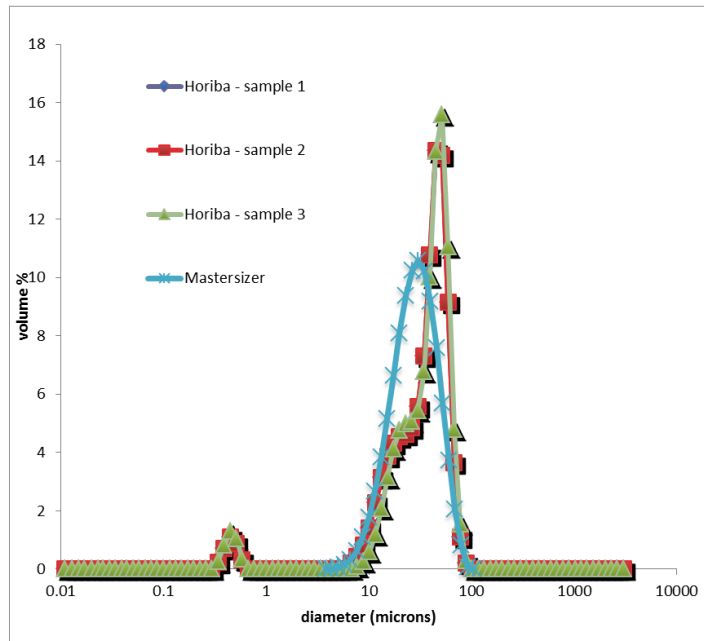
$$\frac{\mathbf{J}_s}{\rho_s} = -(\phi K_c \nabla(\dot{\gamma} \phi) + \phi^2 \dot{\gamma} K_\eta \nabla(\ln \eta)) + f_{\text{hindered}} v_{\text{stokes}} \phi$$

$$\nabla \cdot \mathbf{v} = \frac{(\rho_f - \rho_s)}{\rho_s \rho_f} \nabla \cdot \mathbf{J}_s$$

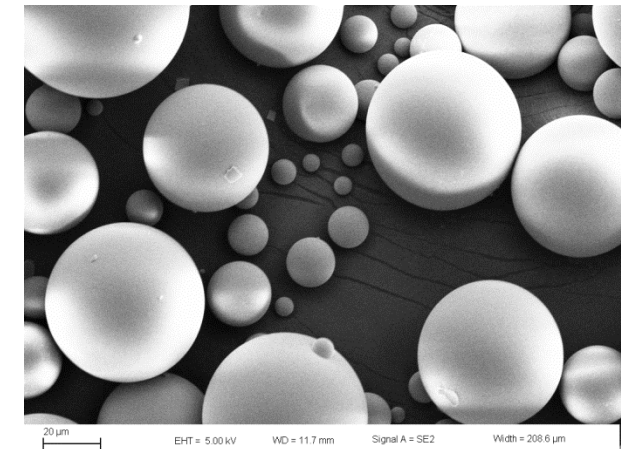
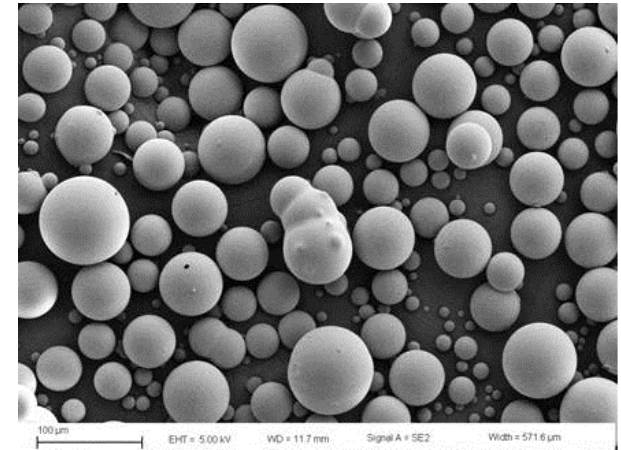
$$\rho C_p \frac{dT}{dt} = -\mathbf{v} \cdot \nabla(\rho C_p T) - \nabla \cdot \mathbf{q} + H(\xi)$$

- Diffusive flux model included in open source finite element code GOMA.
(Phillips et al., Phys. Fluids A 1992; Rao et al, Int. J Numer. Meth. Fluids 2006; Mondy et al. J. Appl. Poly. Sci. 2011; GOMA Research & Development 100 Award 2014)
- Model parameters that must be collected to model flow and track particles for each new encapsulant are circled in red.

3M™ Glass Bubbles (GMB) Characterization: Size



- Two instruments using laser diffraction method (intensity converted to volume distribution based on glass properties)
 - Horiba Laser Scattering Particle Size Distribution Analyzer LA-950
 - Malvern Mastersizer 3000 laser diffraction particle size analyzer
- Laser diffraction on hollow spheres?



3M™ Glass Bubbles (GMB) Characterization: Density

Particle	Type of measure	Density g/cm ³
D32 unseived	Manufacturer	0.32
D32 unsieved	Weigh known volume and estimate max packing (0.68 - 0.80)	0.260 – 0.22
D32-Sieved-65µm-45µm	Pycnometer	0.234 s.d.=0.00027
D32-Sieved-25µm	Pycnometer	0.162 s.d.=0.00046

Hindered settling

$$u_d / u_{Stokes} = f$$

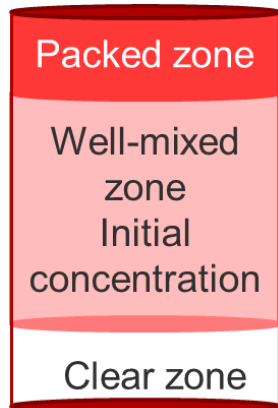
Richardson – Zaki :

$$f = (1 - \phi)^n$$

Acrivos :

$$f = \frac{(1 - \phi)}{\mu_{rel}}, \mu_{rel} = \left(1 - \frac{\phi}{\phi_{max}}\right)^m$$

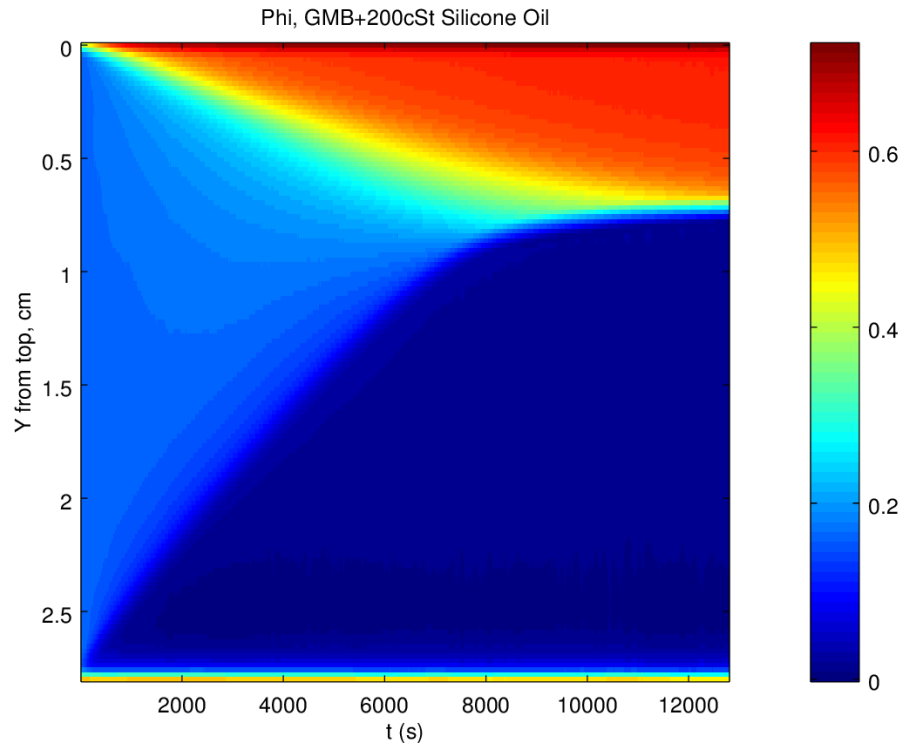
When a suspension settles (or floats), a particle goes slower than the Stokes' velocity of an isolated particle.



A broad distribution of particle sizes leads to diffuse interface between particles and clear zone and a “well-mixed” zone that has changing average particle size

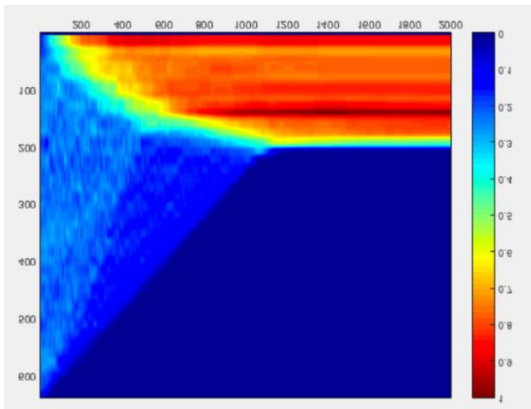
- Suspension of uniform particles will form 3 distinct zones when floating . A sharp front forms between the two extremes and the suspension in the middle remains close to the initial concentration.
- Most studies track the interface velocity to get the hindrance function f .
- NMR allows calculation of discrete phase average velocity (Altobelli & Mondy, 2002)
 - ✓ Do not have to track a clearing front velocity, especially if the front is not sharp because of a broad distribution of particle sizes
 - ✓ Can get same hindered settling function in one experiment, instead of having to test a range of particle concentration

NMR Imaging to Determine Hindrance Function



- Fluid fraction p_c is proportional to NMR signal strength
- 1-D NMR imaging can map the particle concentration with height Y and time t
- For each discrete value of t (Δt on order of 1 minute), we get a measure of p_c at each location y ($\Delta Y=0.05$ cm)

$$\phi = (1 - p_c)$$



Bimodal results 2:1 size ratio from discrete-particle simulation (2000 particles) by the lattice-Boltzmann technique (Ladd 1994; Aidun, Lu & Ding 1998)

NMR Imaging to Determine Hindrance Function

- Continuity of liquid phase:

$$\nabla \cdot p_c u_c = - \frac{\partial p_c}{\partial t}$$

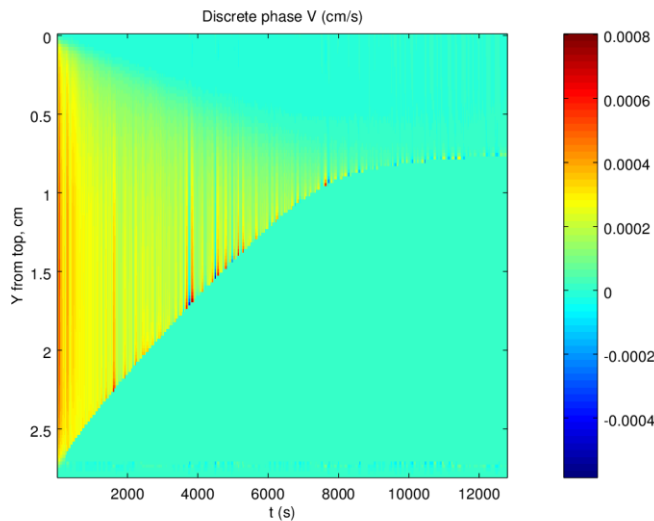
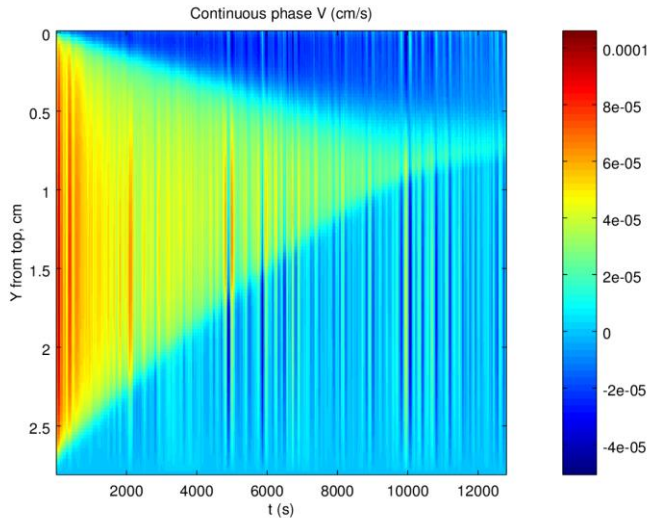
- Assume 1-D flow

$$\frac{\partial u_c}{\partial Y} = \frac{-\left(\frac{\partial p_c}{\partial t}\right) - u_c \left(\frac{\partial p_c}{\partial Y}\right)}{p_c}$$

- Calculate partials from NMR data
- At each time step, evaluate $\partial u_c / \partial Y$ (RHS) for each Y and integrate starting at $Y=0, u_c=0$ to give u_c at next time step

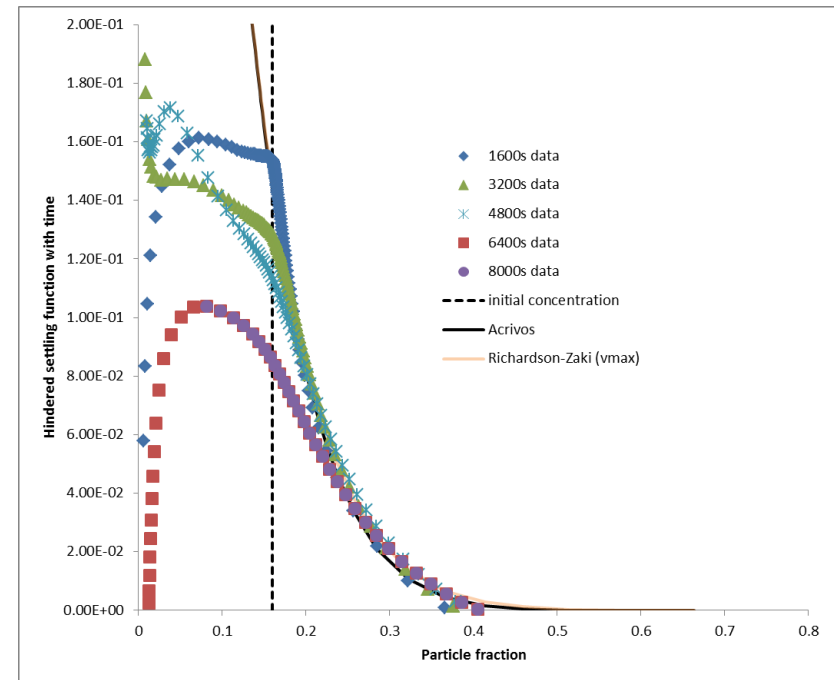
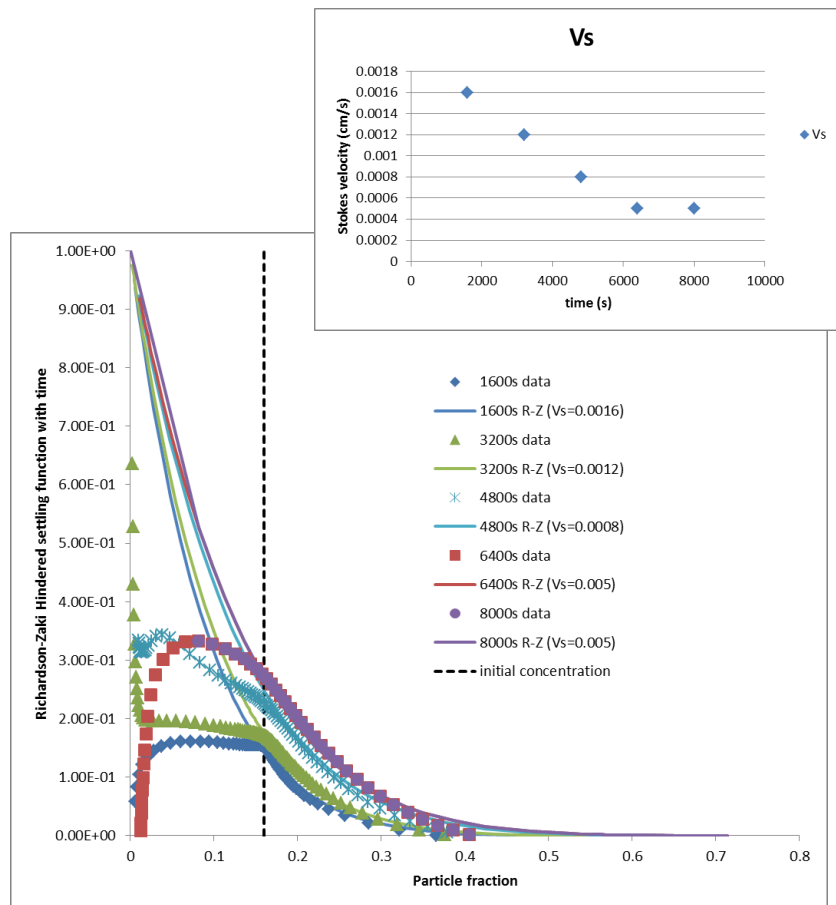
- Discrete phase velocity is related (upward flux of one phase must equal downward of the other):

$$u_d = - \frac{u_c (1 - \phi)}{\phi}$$



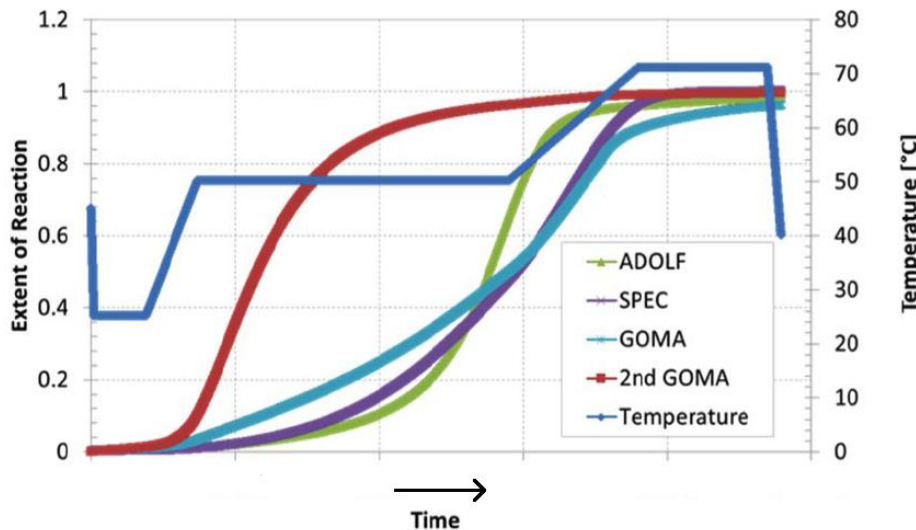
NMR Imaging to Determine Hindrance Function

- Well-behaved data above the initial concentration
- Can either :
 - allow the fit to change as the particles size segregate (left) or
 - fit to largest particles and miss the later time behavior (right)



Polymerization Kinetics

- Differential Scanning Calorimetry used to determine heat of reaction and extent of reaction in isothermal cures (by integrating heat flow, extent of reaction at any time could be estimated)
- Model forms yield very different extent of reaction with time
- Various kinetic models fit to data (Epon 828 cured with DEA): SPEC model best fit with and without added GMB

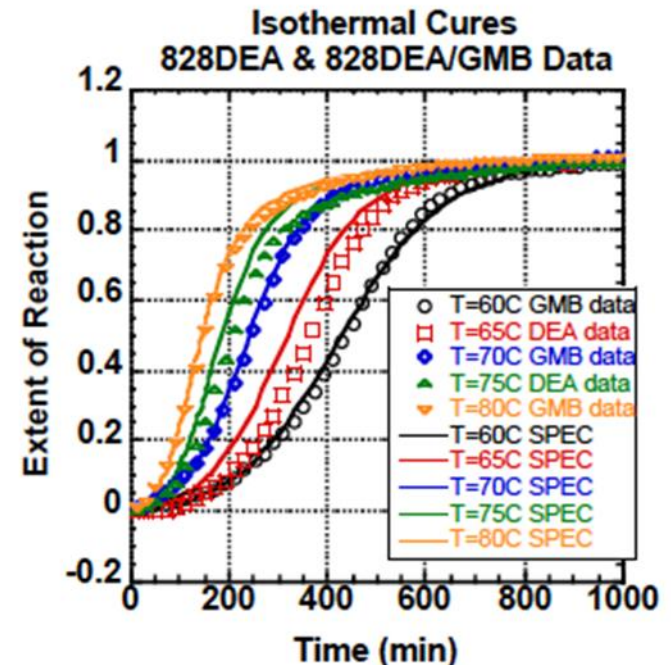


SPEC:

$$\frac{dx}{dt} = \frac{k_o \exp(-E_a/RT)}{(1 + wa_{shift})^\beta} (b + x^m)(1 - x)^n$$

$$n = n_1 + n_2 \tanh(n_3 \{T - n_4\})$$

Erin Karasz



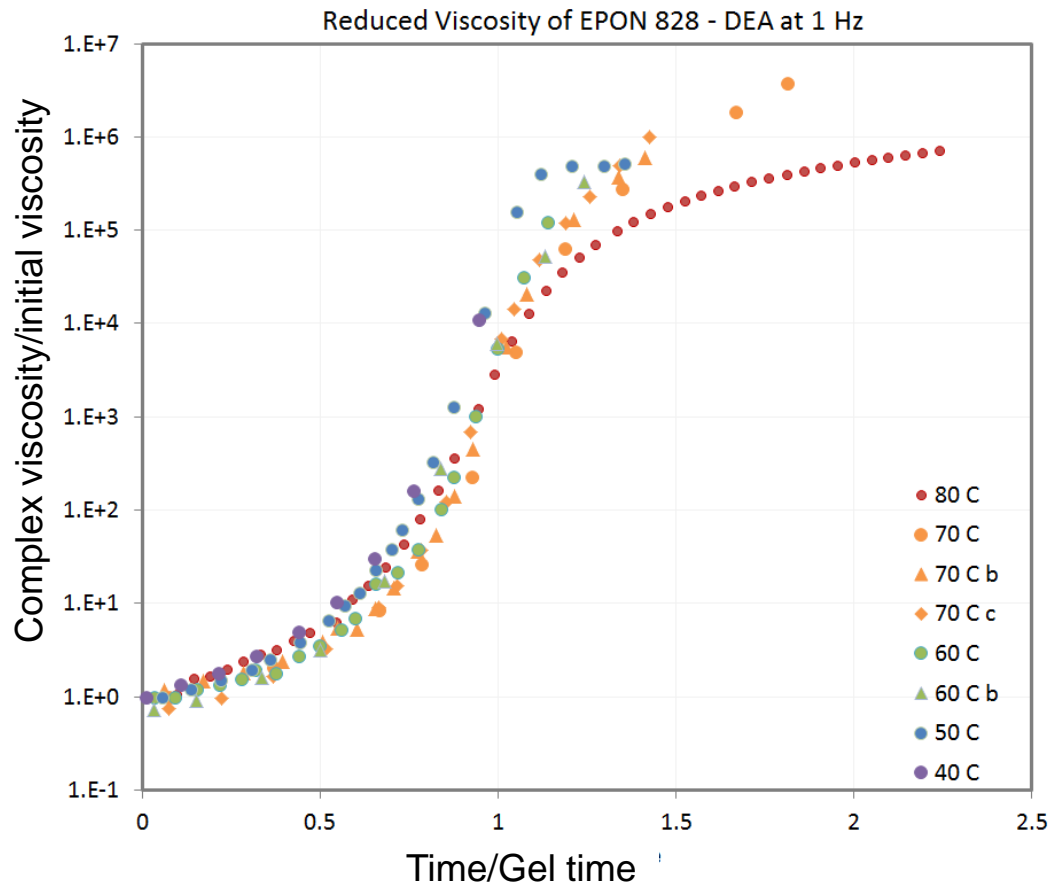
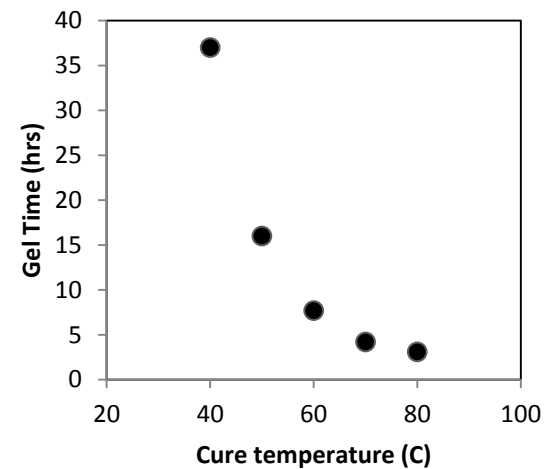
We assume $w=0$ for now and no vitrification is seen in the liquid phase

Bob Chambers, Doug Adolf

Rheology of Continuous Phase: EPON 828 + DEA During Curing

Measured with ARES rheometer using disposable parallel plates (25mm). Isothermal experiment

Challenging due to length of time for experiments, sudden high stresses at cure

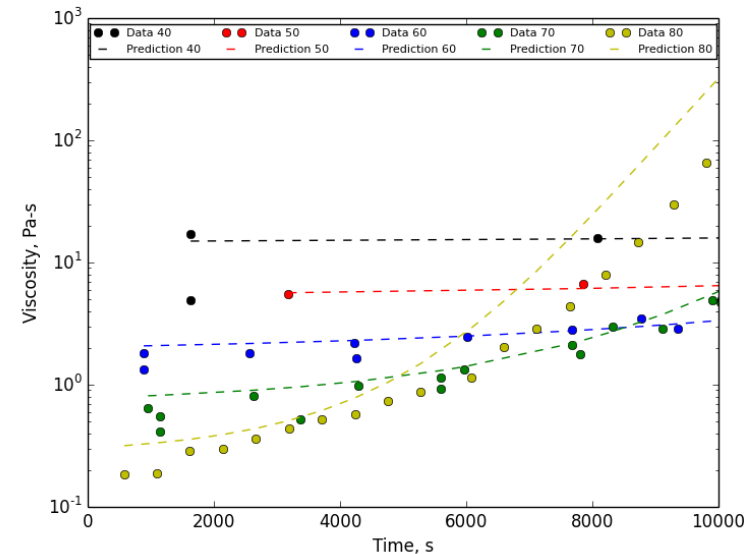
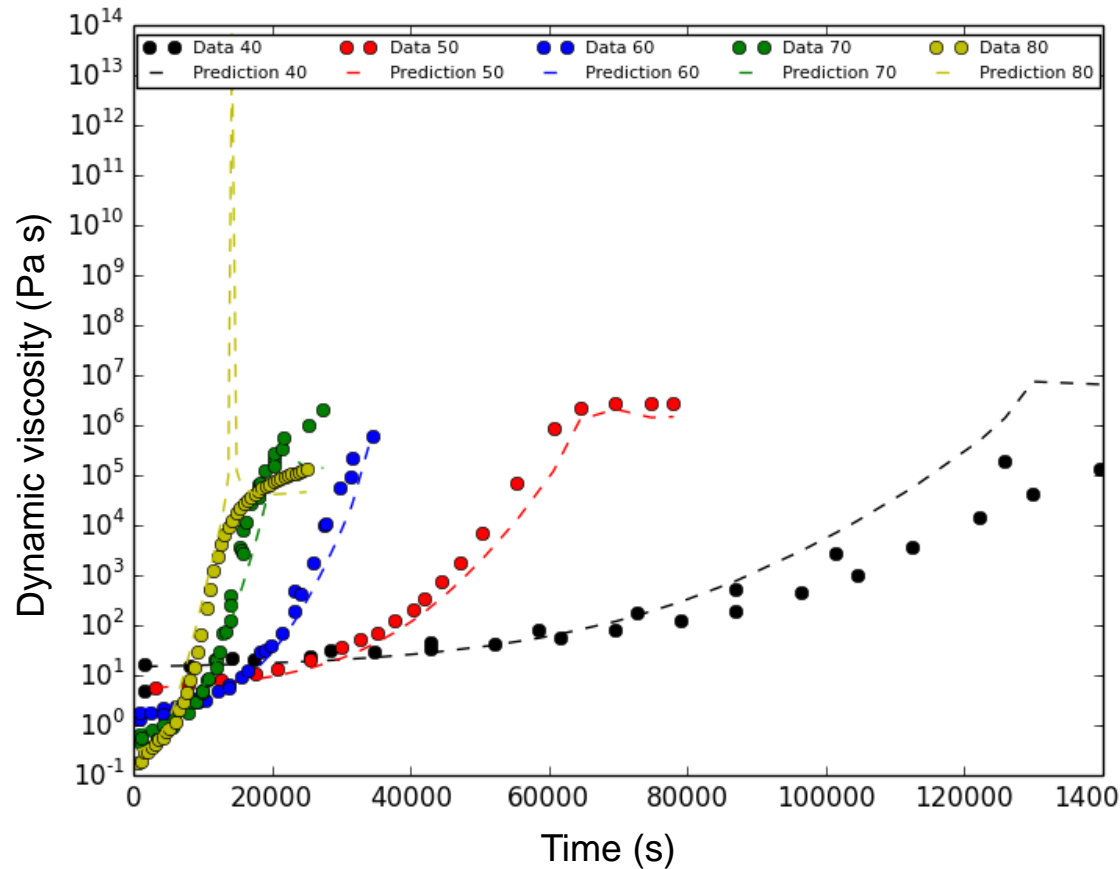


- Modulus vs. frequency collected at each time point throughout cure
- Initial viscosity and final shear modulus are both dependent on cure temperature
- Collapse of data when scaled by gel time and initial viscosity at that temperature.

Fit to New Curing Rheology Data (Epoxy Only)

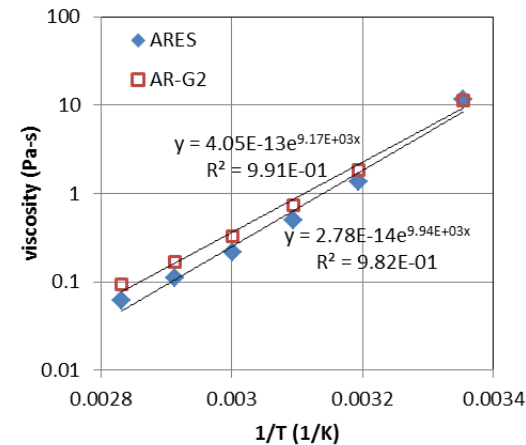
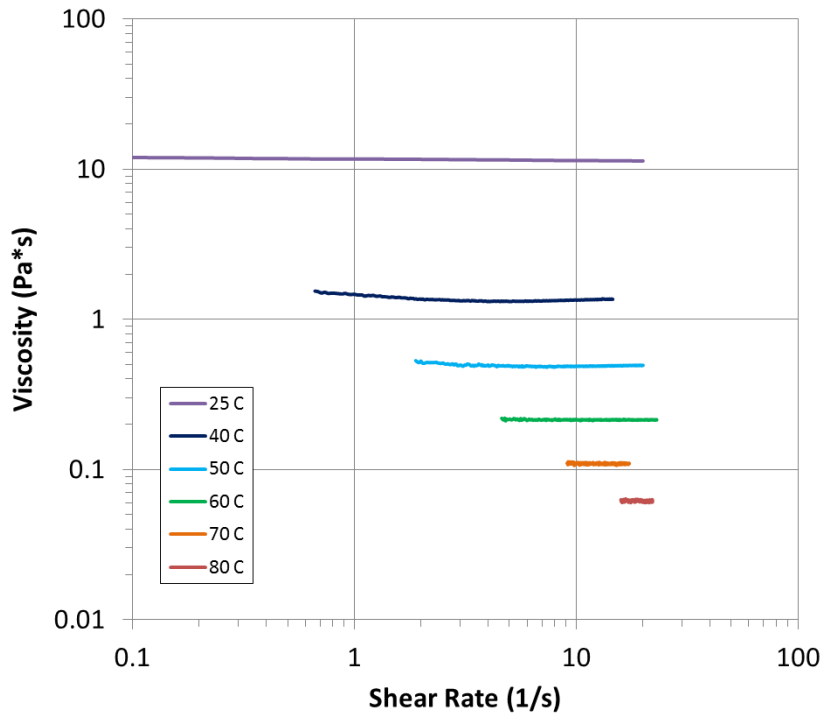
$$\eta = \eta(\xi, T) = \eta_0(T_g) \left(1 - \left(\frac{\xi}{\xi_c} \right)^m \right)^{-p} 10^{\frac{c_1(T-T_g)}{c_2+T-T_g}} \quad T_g = \frac{T_{g0}}{1 - A\xi}$$

$$\begin{aligned} \eta_0 &= 2.016\text{e-}01 \text{ Pa-s} & m &= 2 \\ C_1 &= 40.13 & p &= 4/3 \\ C_2 &= 999.3 \\ A &= 0.2336 \\ \xi_g &= 0.8137 \\ T_{g0} &= 357.6 \text{ K} \end{aligned}$$



Rheology of Filled (Non-Curing) Systems: Neat Epoxy

- Measured neat Epon, then filled Epon at several temperatures
- Neat Epon: Viscosity was measured using two different rheometers using two different geometries. (parallel plates expected to be less accurate, but has to be used for curing system)
 - ARES = 25 mm parallel plates
 - AR-G2 = Couette

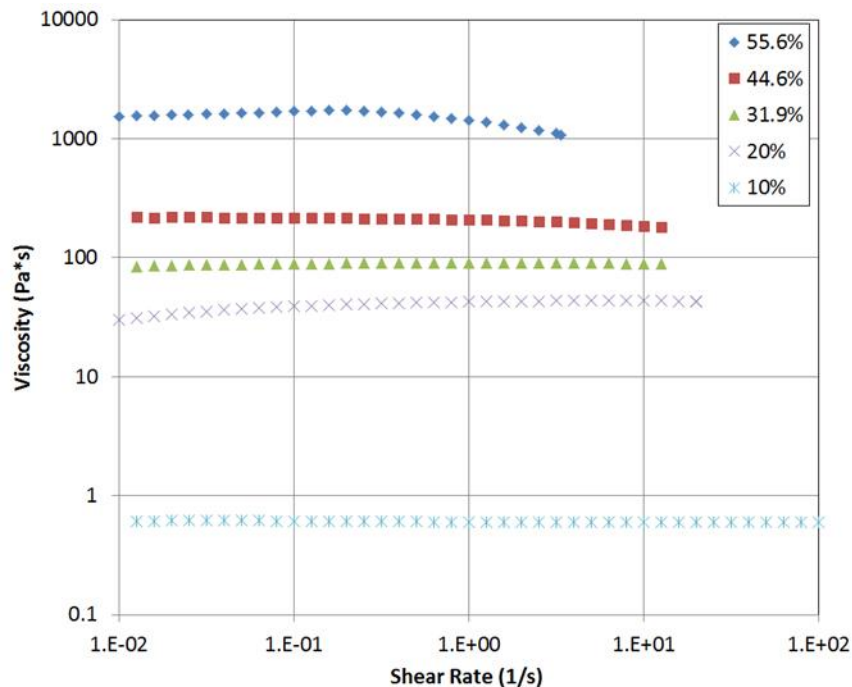


Viscosity follows Arrhenius type behavior with temperature, as expected.

No dependence on shear rate.

Rheology of Filled (Non-Curing) Systems: Behavior as a Function of GMB Volume Fraction

- Measured vs. shear rate using AR-G2 rheometer, double gap cylinder geometry
 - Particle migration less of a concern
- Viscosity is approximately Newtonian



Epon + GMB

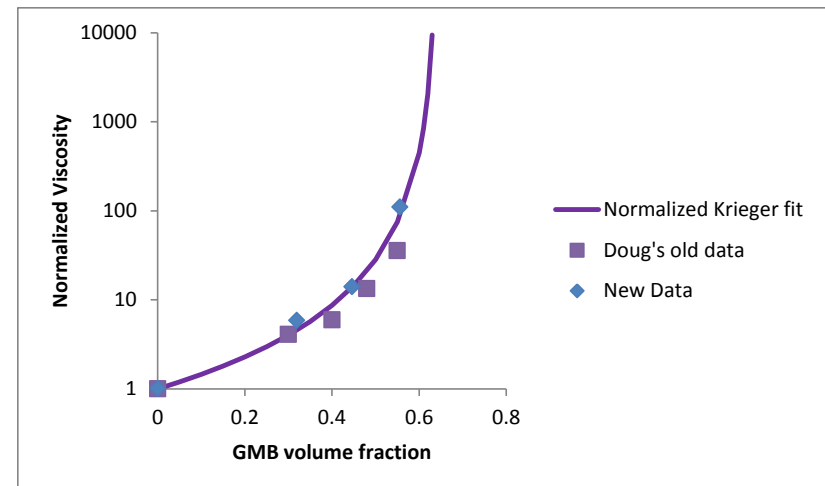
Parameters for
New Krieger model

$$\phi_{\max} = 0.64$$

$$n = 2.2$$

$$\eta = \eta_0 \left(1 - \frac{\phi}{\phi_{\max}} \right)^{-n}$$

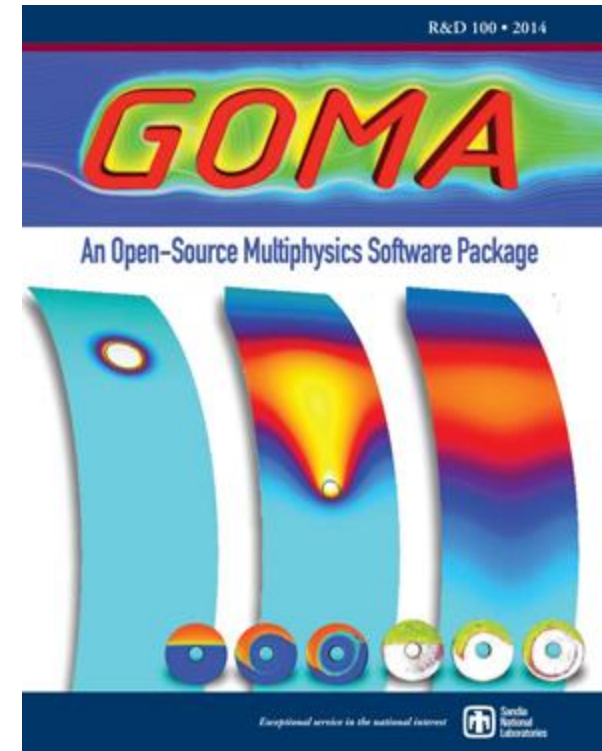
In model η_0 is taken to be the curing continuous phase viscosity



Doug Adolf

Numerical Method & Issues

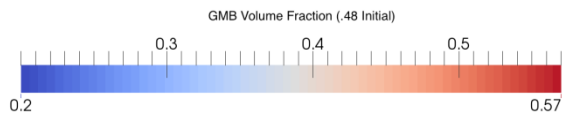
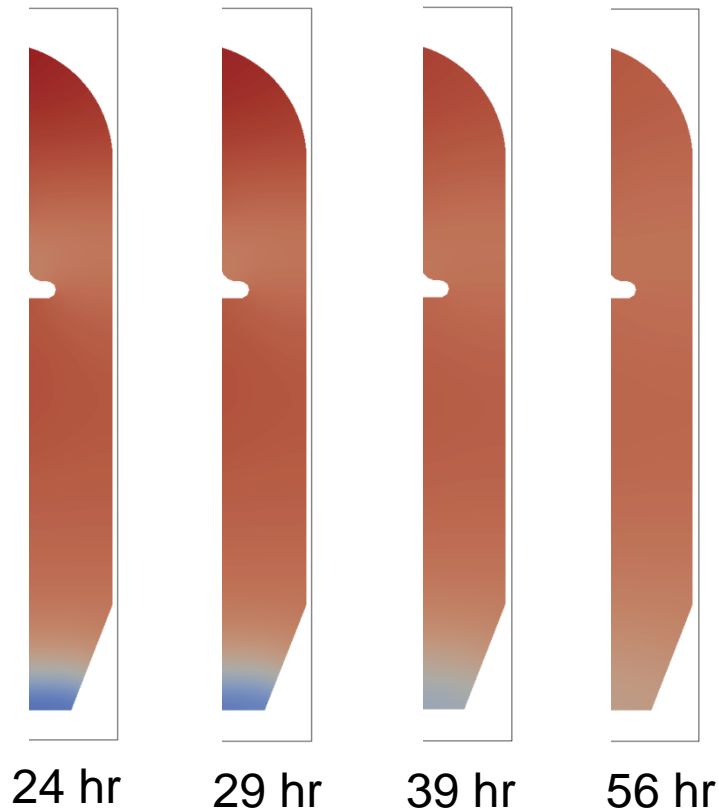
- Galerkin Finite element method used for spatial discretization with biquadratic velocities, bilinear pressure, shear-rate and concentration
- Finite difference method used for temporal discretization
- Taylor-Galerkin 2nd order method used for species equations time integration
- Fully coupled solution algorithm using Newton's method
- Pressure stabilization (GLS) is used to enable use of iterative solvers, which reduces solution times (Hughes et al., 1990)
- Parallel implementation based on MPI
- Boundary conditions are generally simple such as no slip or specified heat flux
- Shear-rate invariant must be interpolated as a variable
- Kinematic shocks form as suspension settles forming three phases: fluid phase, a particle phase at maximum packing and a suspension phase
- Fluid phase/suspension interface susceptible to oscillations that grow in time



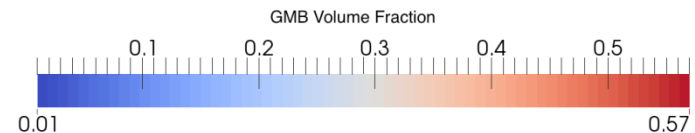
Example Numerical Results

Process/geometry changes can be examined more rapidly with modeling

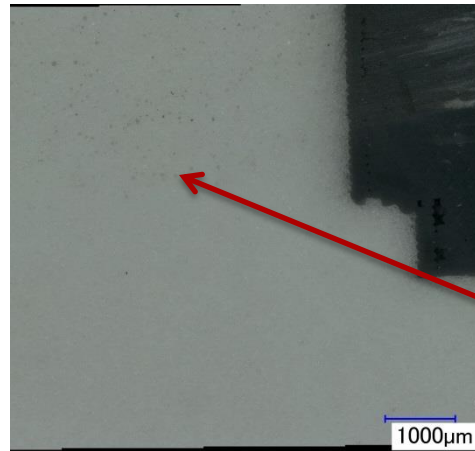
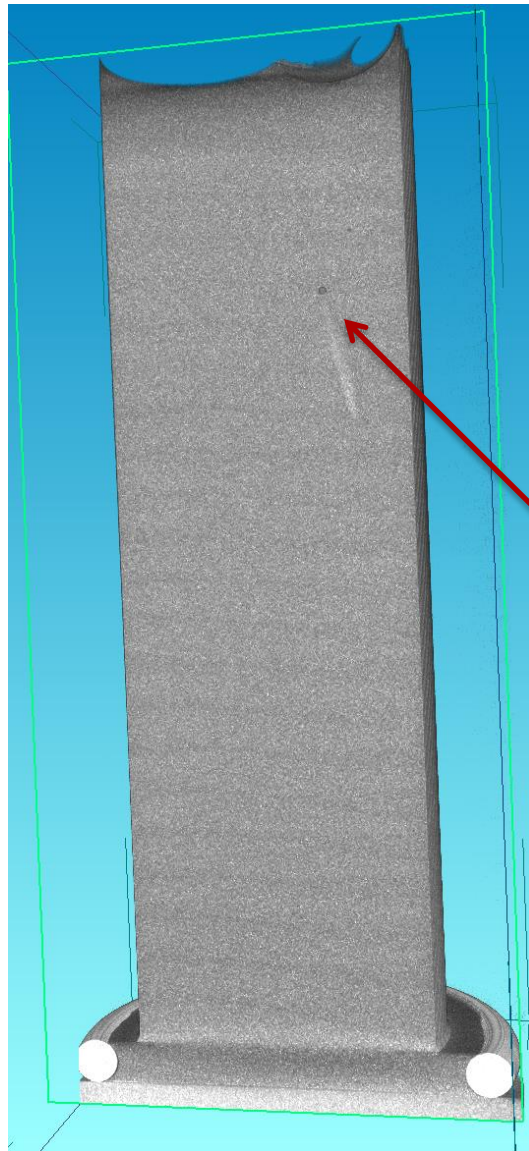
Particle migration minimized in 56
hour cure



Varied volume fraction: 0.3 to 0.48
Used 27 hour cure profile



What's Missing?



Particles size segregate during cure

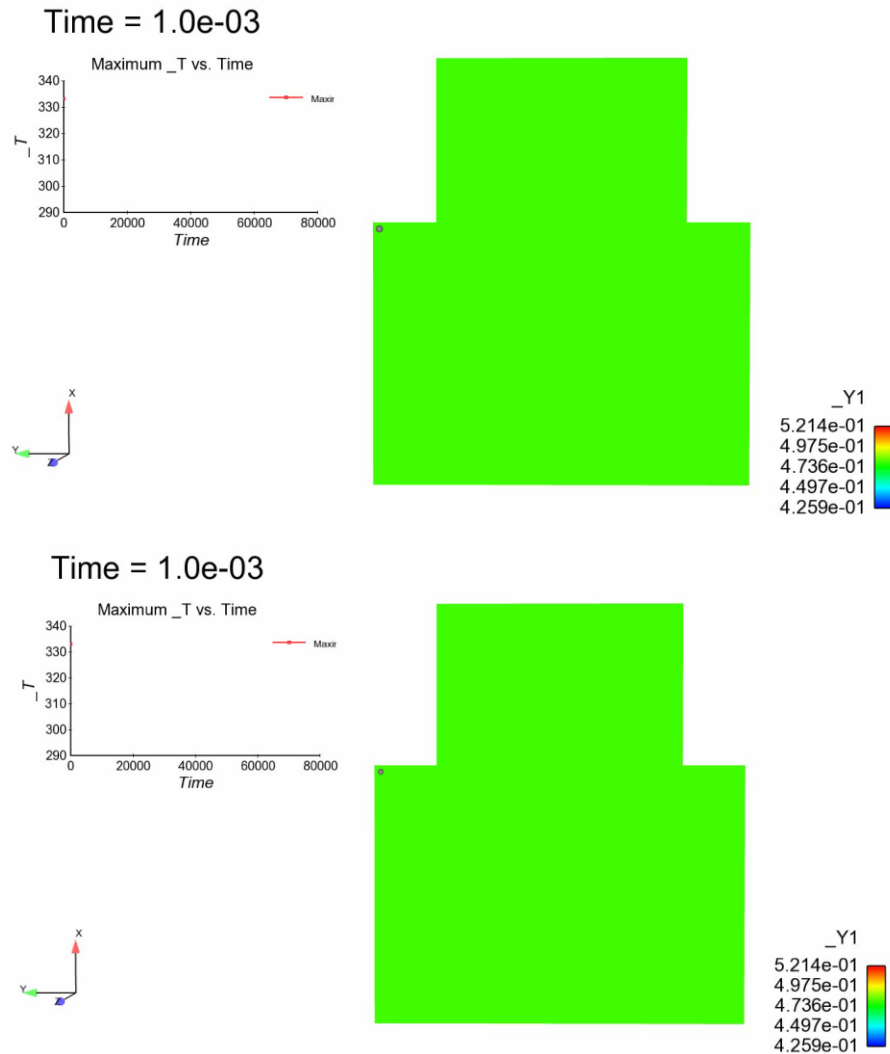
Larger particles can be seen at top

Large bubble formed when pouring into a cylindrical mold, rises during curing, leaving an epoxy-rich region in its wake.

bright => higher density => higher epoxy content
dark => higher particle content (or void)

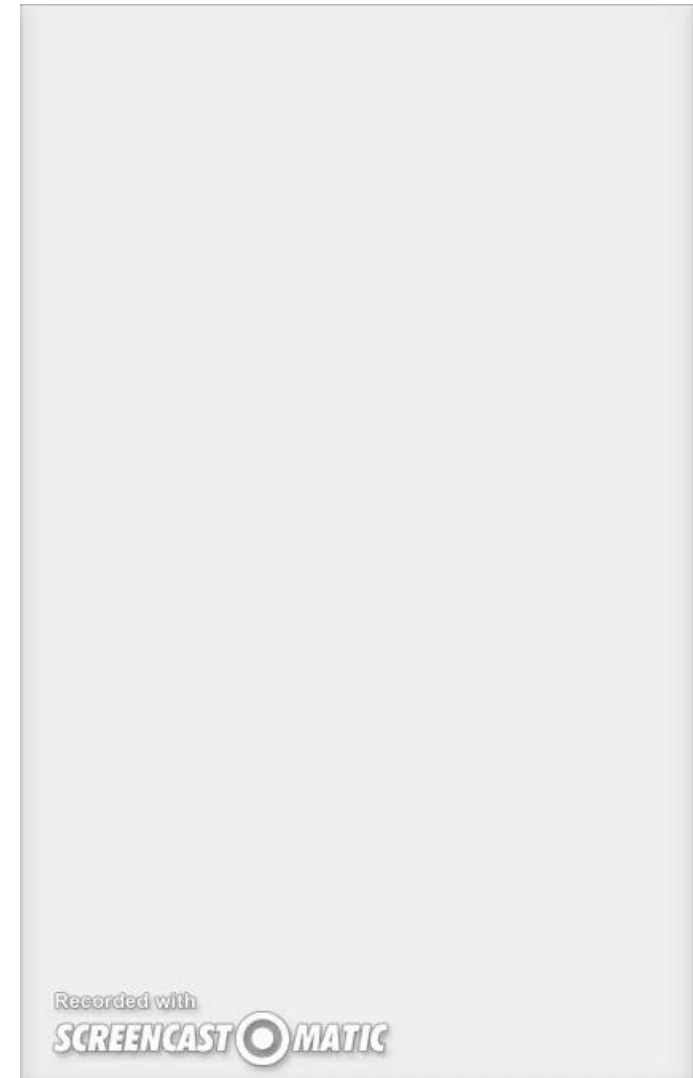
Summary

- Goal is to predict distribution of filler particles as a result of processing time and temperature changes
- Finite element model developed to help optimize processing conditions
- Experiments performed to determine model parameters. Improvements needed.
 - Particle characterization – sizes and densities
 - Hindered settling function
 - Polymer kinetics
 - Rheology – effects of particle fraction and extent of cure



Next Steps

- Improve hindered settling function description
 - Better particle characterization
 - Capture effects of size distribution: Can we capture size segregation?
 - Examine effects of possible surface interactions
- Improve curing and rheology dependence
 - Improve DEA-Epon curing model with new DSC and IR data
 - Extend rheology measurements to longer times
 - Capture change of curing mechanism at higher temperatures
 - Improve T_g model using Di Benedetto form: Vitrification is seen in rheology data!
 - Measure rheology of filled, curing systems – do we get expected behavior?
- Run 3D models
- Validate against concentration, temperatures in various geometries
- Enrich continuum model with particle migration behavior of multimodal size distributions
 - Sub-grid models to understand more complex behavior: e.g., discrete-particle simulation.
 - Does mold-filling create initial particle concentration/size gradients?



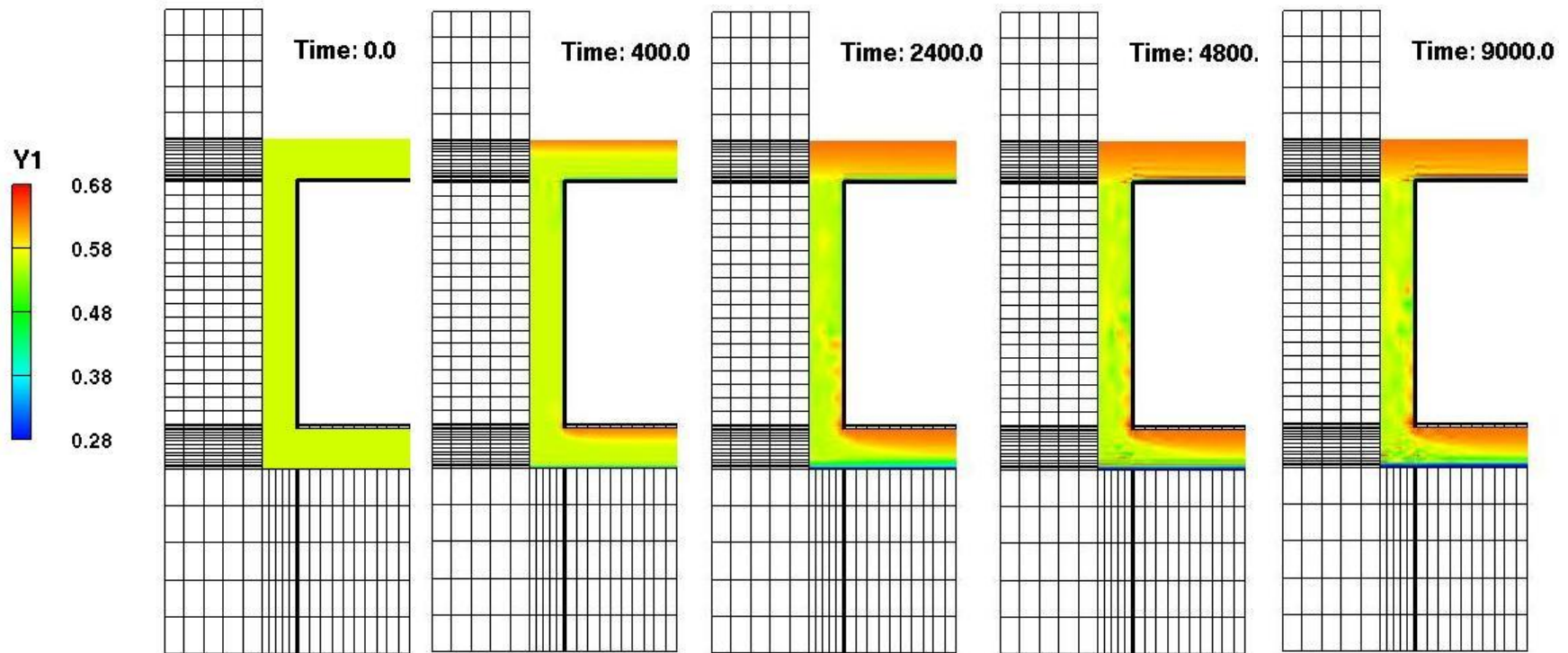
Discrete-particle simulation by the lattice-Boltzmann technique (Ladd 1994; Aidun, Lu & Ding 1998) -- Morris & Pednekhar

Thank you

Job Opening ID: 651319

Posting Title: Thermal Battery Research Postdoctoral Appointee

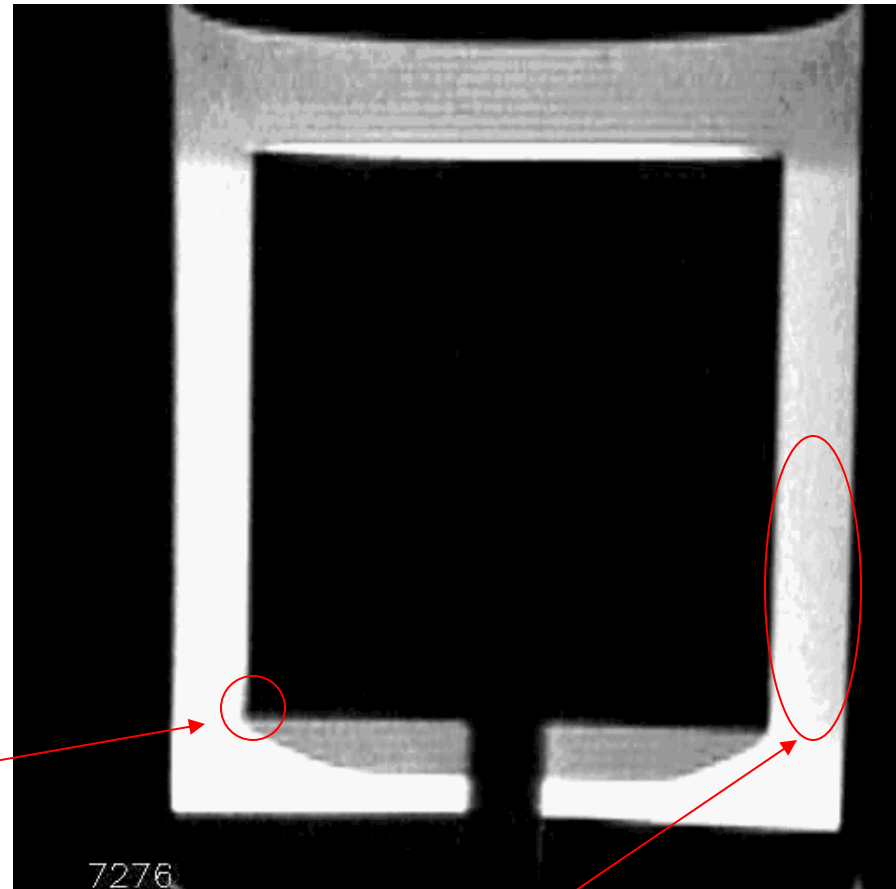
GMB Flotation as a Function of Time for Curing Epoxy



Model shows instabilities that lead to inhomogeneities in particle concentration

GMB Flotation as a Function of Time for Model System

NMR data does not give enough resolution to distinguish possible “blobs” from instabilities



High particle region curving around lip?

Mottled area real?