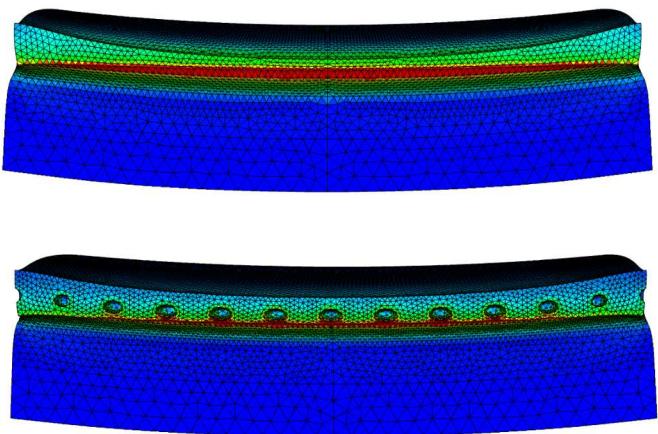
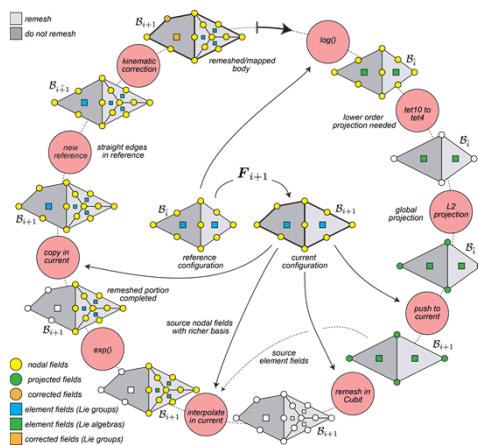
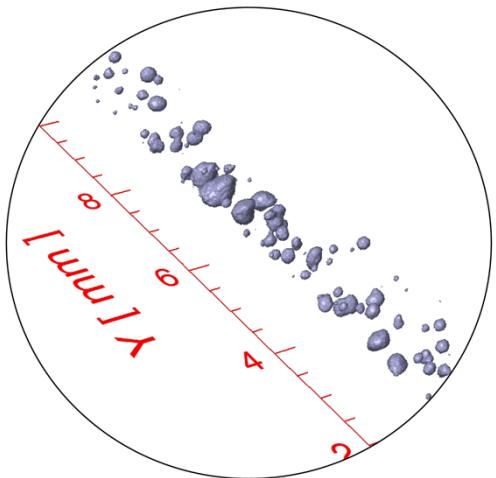


Exceptional service in the national interest



Resolving the evolution of pore structures in 304-L laser welds

J. Foulk, M. Veilleux, J. Emery, J. Madison, H. Jin, J. Ostien, A. Mota
 USNCCM 13, San Diego, July 30, 2015



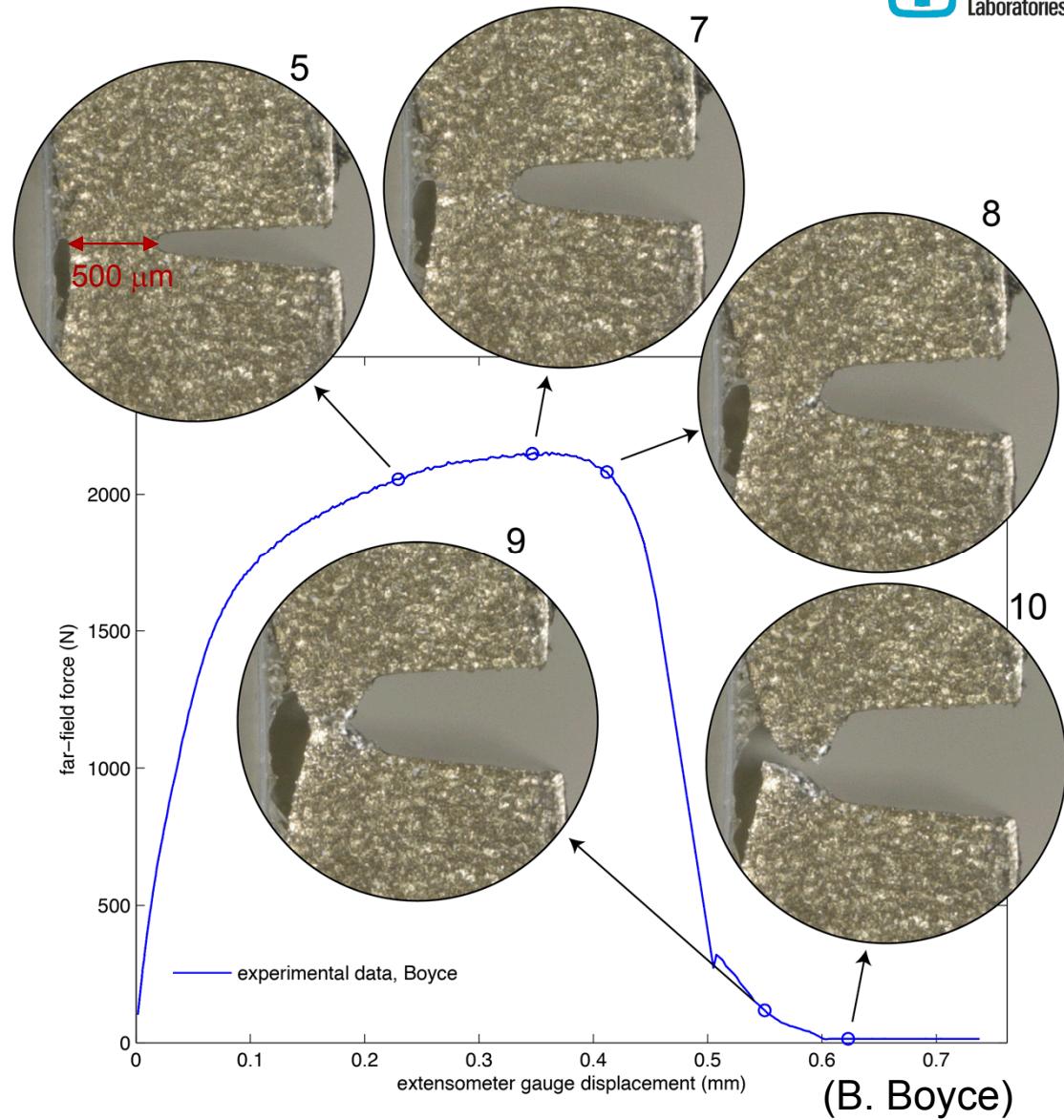
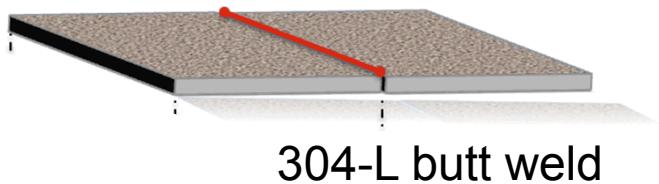
Sandia National Laboratories is a multi-program laboratory managed and operated by Sandia Corporation, a wholly owned subsidiary of Lockheed Martin Corporation, for the U.S. Department of Energy's National Nuclear Security Administration under contract DE-AC04-94AL85000. SAND NO. 2011-XXXX

Foundational engineering problem – laser welds

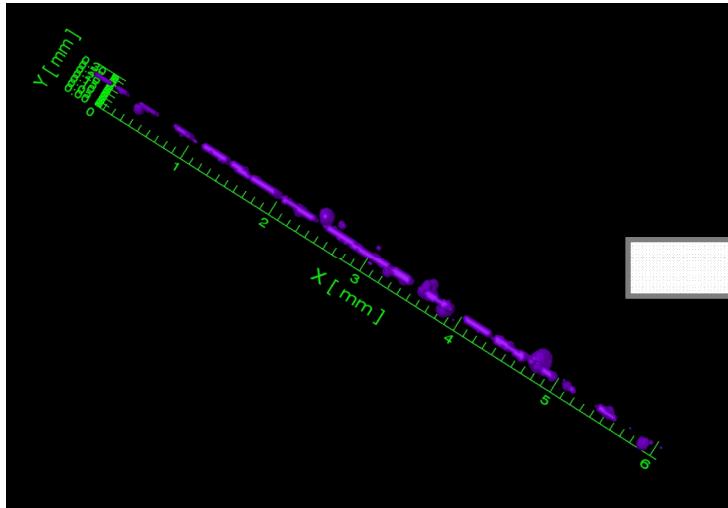
Surface observations indicate that the failure of 304-L is a primarily a necking process. Interrupted testing will determine the role of crack initiation.

Hypothesis: Pore size and distribution can aid the necking process and crack initiation

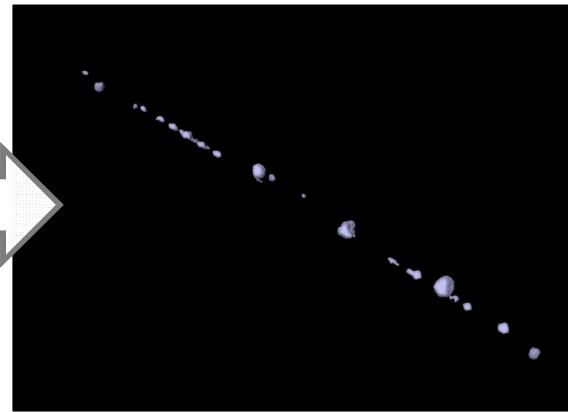
- μ -CT needed to probe initial and interrupted pore structures
- Remeshing/mapping needed to resolve the evolution of pore structure
- Homogenization not applicable



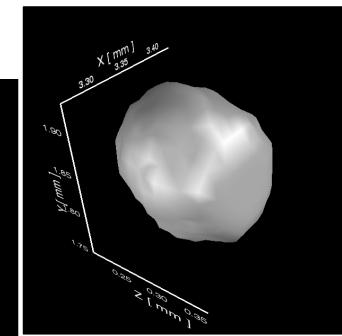
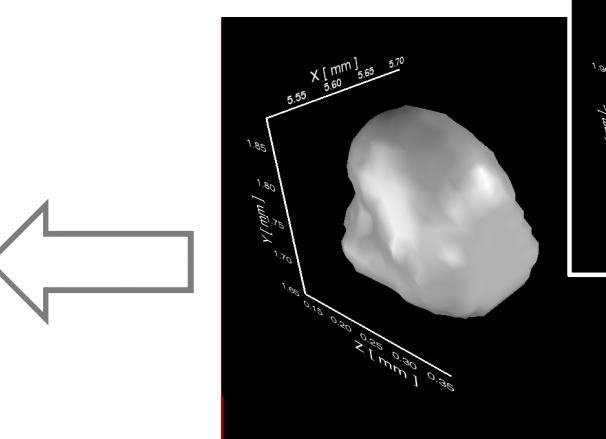
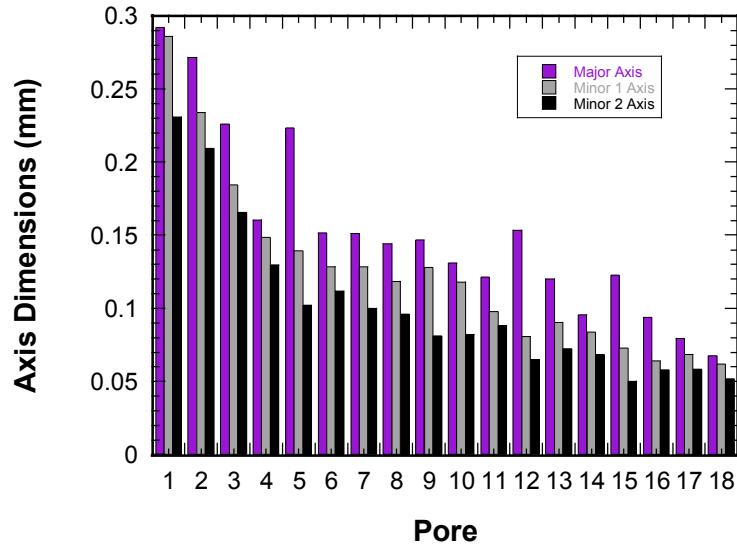
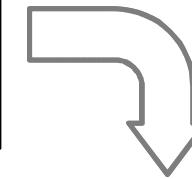
Pores large relative to the ligament – homogenization n/a



μ -Computed Tomography

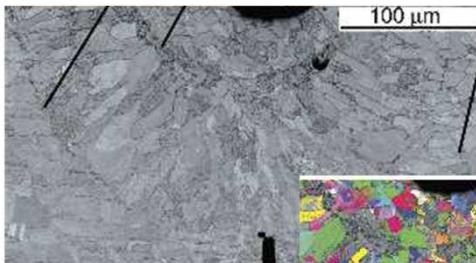


Magnification: 9X
Voxel size: 14 μ m
Energy: 130 keV

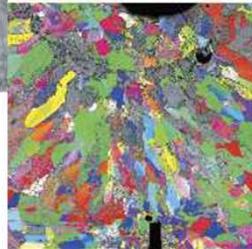


J. Madison, L. K. Aagesen, "Quantitative Characterization of Porosity in Laser Welds of Stainless Steel" *SCRIPTA MATERIALIA* (2012)

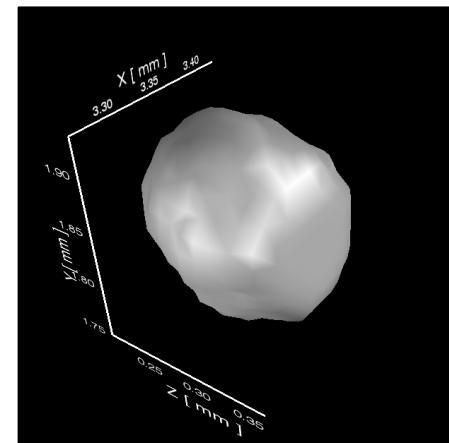
Do the pores dominant the deformation process?



(Boyce & Robino)



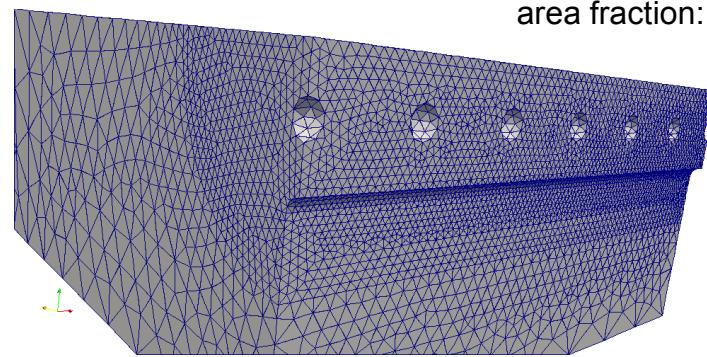
*What elements of microstructure
dominate the load-bearing
capacity?*



(Madison)

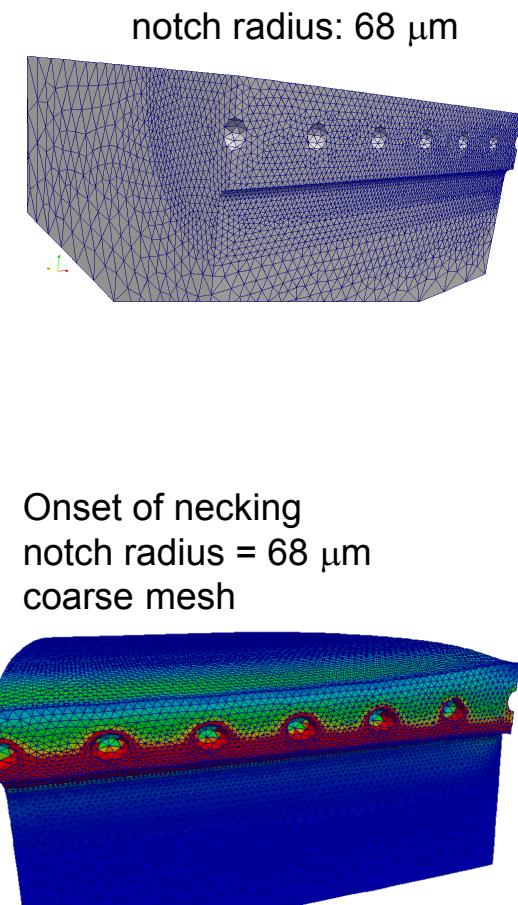
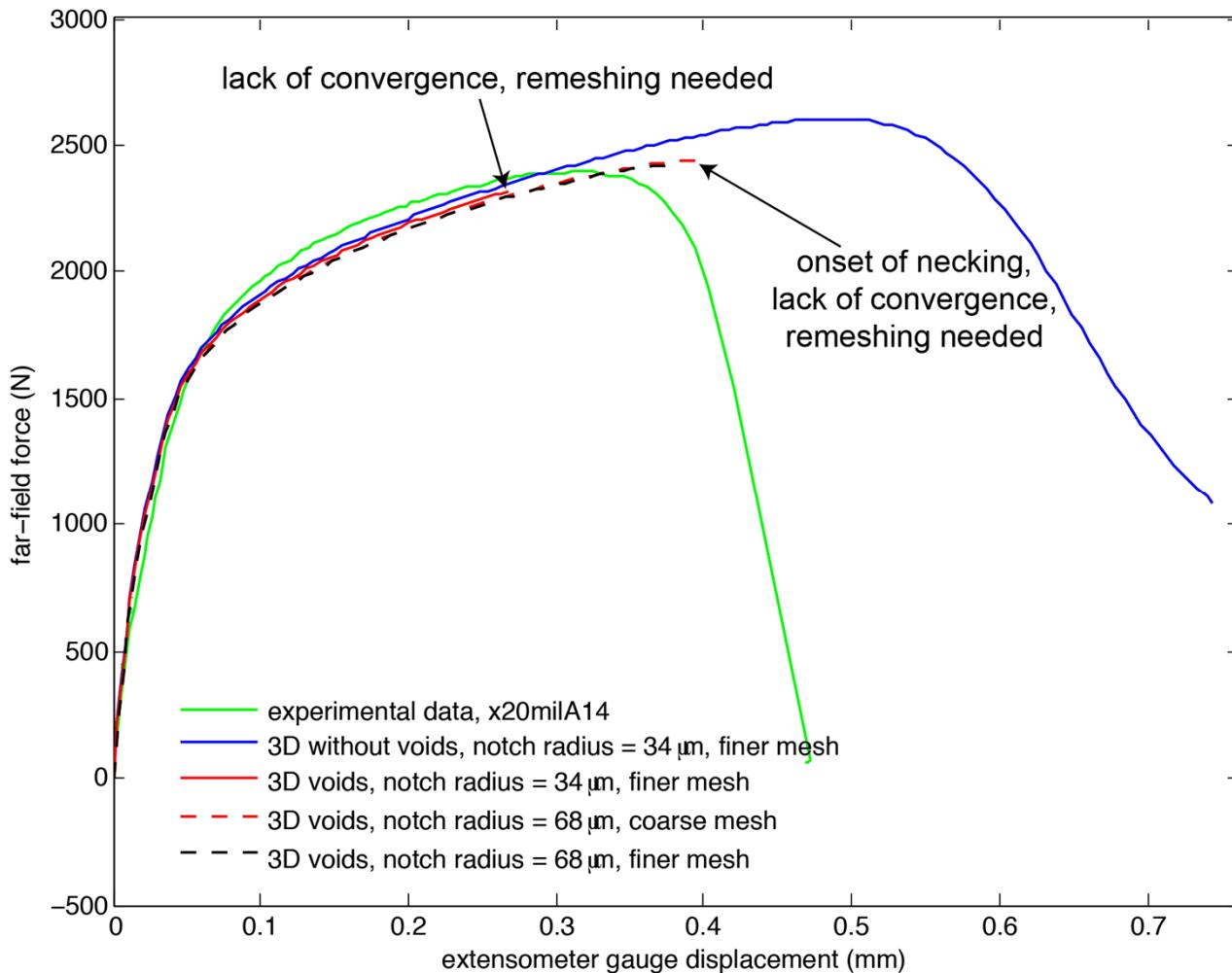
- We hypothesize that pores are the dominant microstructural feature
- We adopt J_2 plasticity for both the base material and the weld
- We have lumped dislocation structures, deformation twinning, and martensitic phase transformations into a phenomenological model for hardening and recovery

sheet thickness: 1.6 mm
ligament length: 508 μm
pore diameter: 150 μm
area fraction: 0.066



NOTE: Unlike experiments, simulation can systematically increase complexity. Pores first.

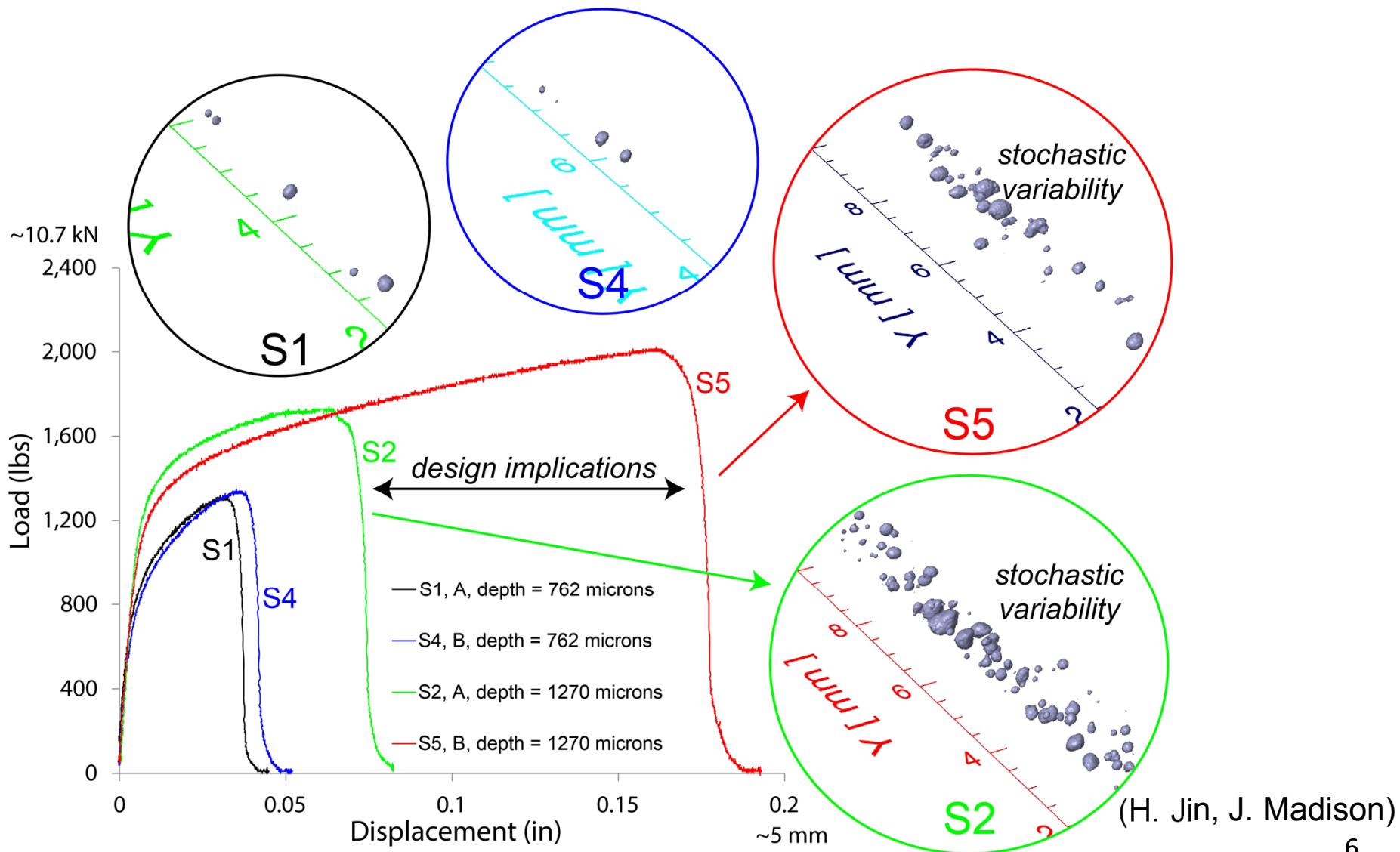
Initial efforts w/pores problematic – remeshing needed



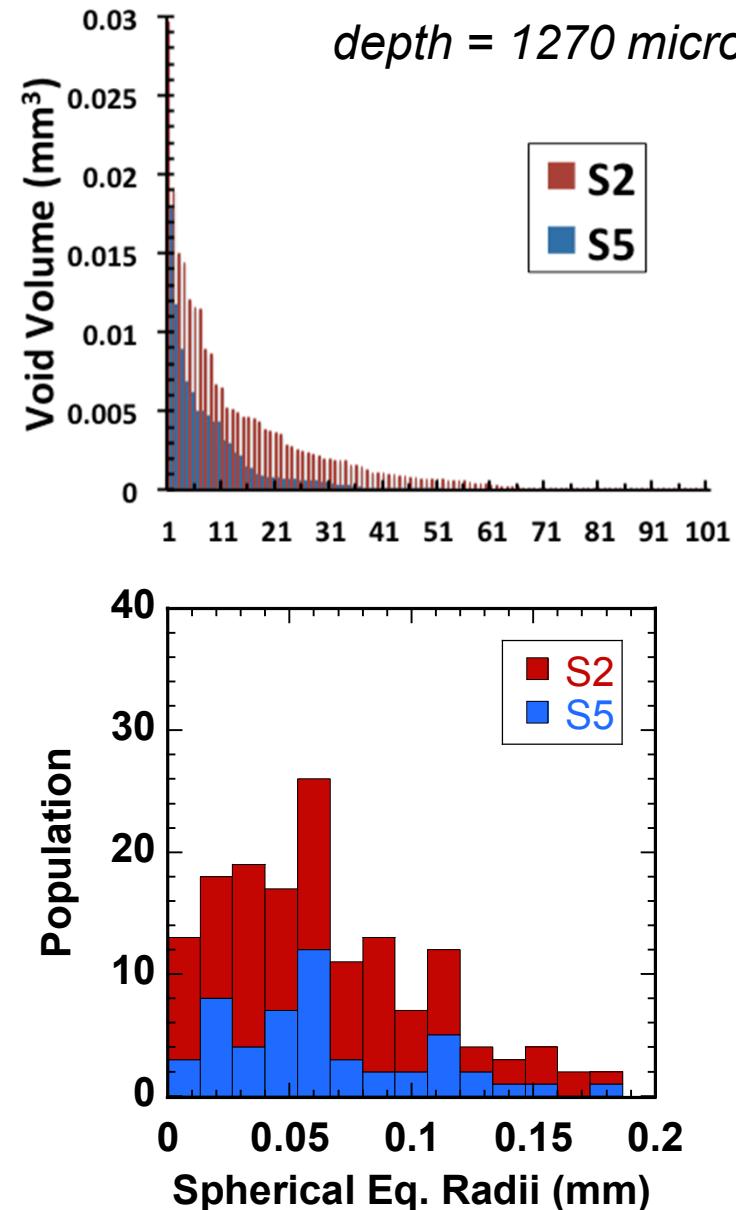
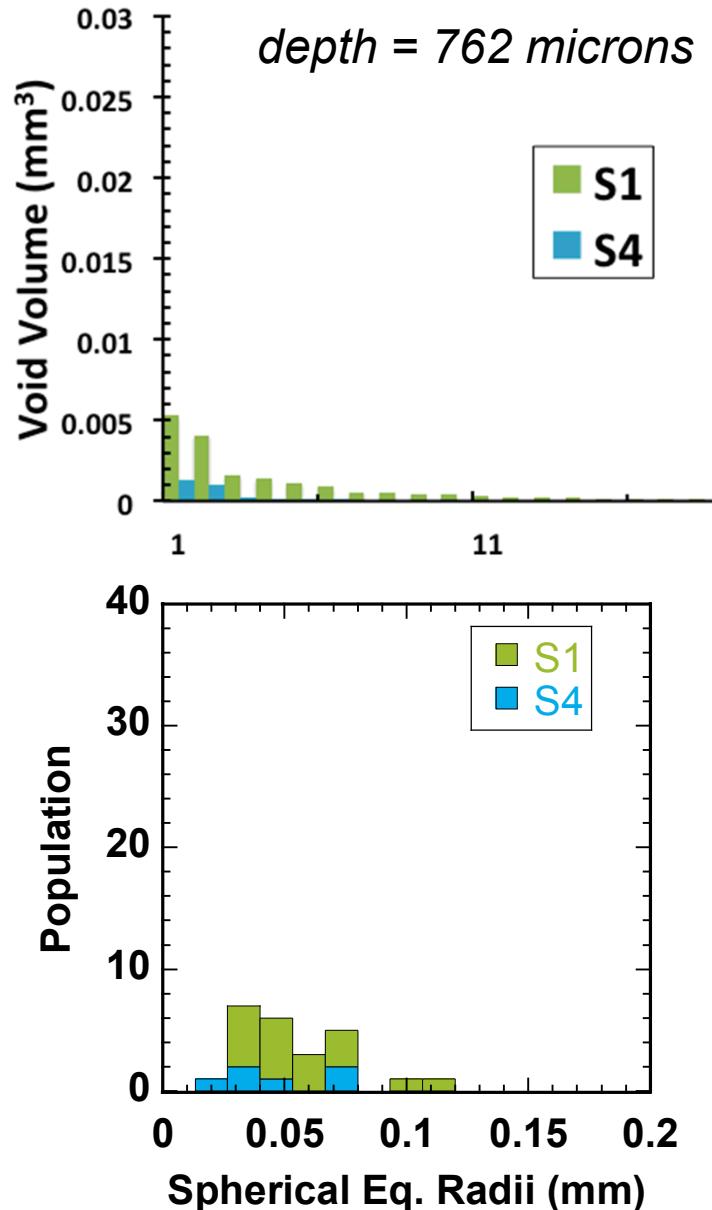
NOTE: Same constitutive model employed for cases with and without voids

Deeper-penetration welds provide additional motivation

Weld schedule impacts porosity. Porosity impacts performance.

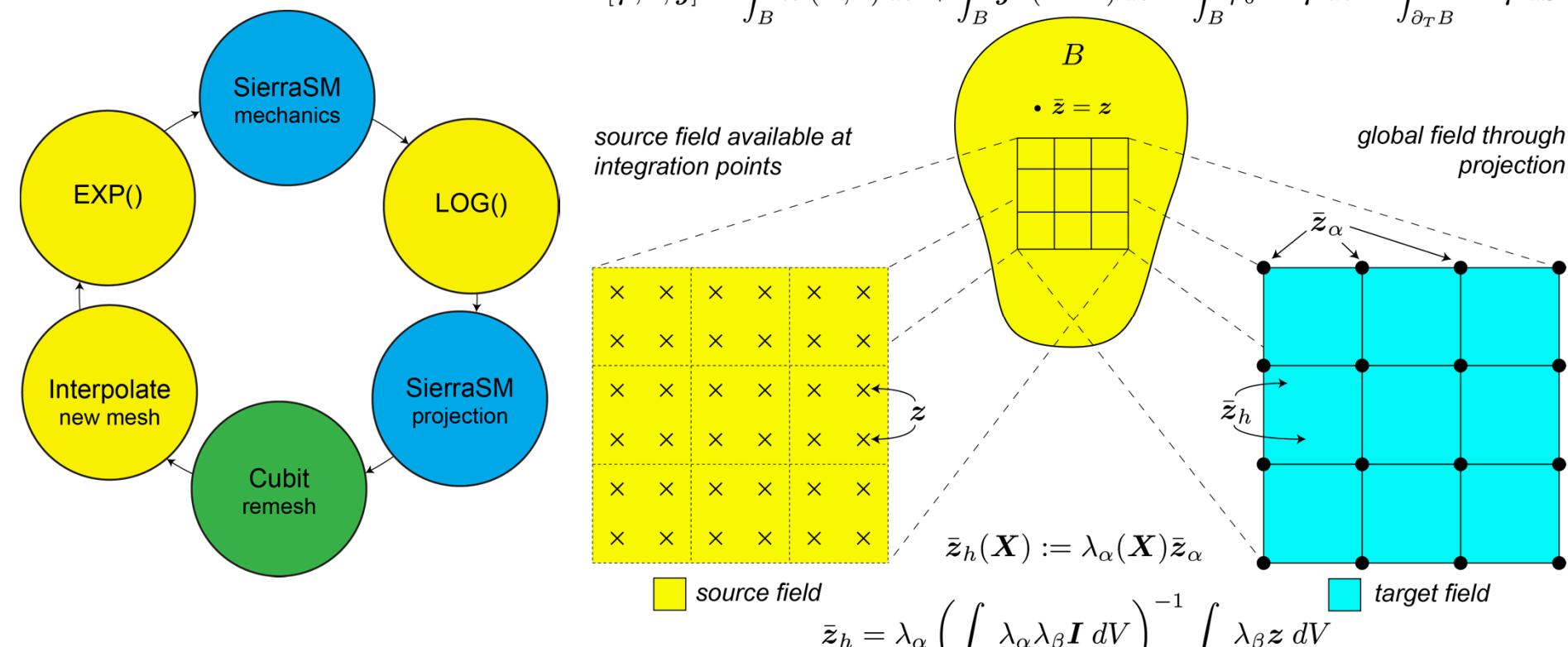


Pore size/distribution varies with weld schedule



Our approach: mapLL (L_2 + Lie Group/Algebra)

$$\Phi[\varphi, \bar{z}, \bar{y}] := \int_B W(\mathbf{F}, \bar{z}) \, dV + \int_B \bar{y} \cdot (\bar{z} - z) \, dV - \int_B \rho_0 \mathbf{B} \cdot \varphi \, dV - \int_{\partial_T B} \mathbf{T} \cdot \varphi \, dS$$



- The variational principle naturally yields an optimal, L_2 projection
 - The spaces of variables (Lie algebra, Lie Group) are honored through $\log()$ and $\exp()$
 - Advocated by Mota, et. al., Computational Mechanics, 2013
- Past works: Ortiz and Quigley (1991), Radovitzky and Ortiz (1999), Rashid (2002), Jiao and Heath (2004)

New tetrahedral element technology – gradient operator

Motivated by prior work of Thoutireddy, et. al., IJNME (2002)

$$\bar{\mathbf{F}}(\mathbf{X}) := \bar{\mathcal{B}}_a(\mathbf{X}) \mathbf{x}_a$$

$$\bar{\mathcal{B}}_a(\mathbf{X}) := \lambda_\alpha(\mathbf{X}) \left[\int_{\Omega} \delta_{ik} \lambda_\alpha(\mathbf{X}) \lambda_\beta(\mathbf{X}) \, dV \right]^{-1} \int_{\Omega} \lambda_\beta(\mathbf{X}) \frac{\partial N_a(\mathbf{X})}{\partial X_J} \, dV \mathbf{e}_i \otimes \mathbf{E}_J \otimes \mathbf{e}_k$$

$$\bar{\mathcal{B}}_a(\boldsymbol{\xi}) = \lambda_\alpha(\boldsymbol{\xi}) \left[\int_{\Omega_\xi} \delta_{ik} \lambda_\alpha(\boldsymbol{\xi}) \lambda_\beta(\boldsymbol{\xi}) \, dV_\xi \right]^{-1} \int_{\Omega_\xi} \lambda_\beta(\boldsymbol{\xi}) \frac{\partial N_a(\boldsymbol{\xi})}{\partial \xi} \, dV_\xi \left(\frac{\partial \boldsymbol{\xi}}{\partial X_J} \right) \mathbf{e}_i \otimes \mathbf{E}_J \otimes \mathbf{e}_k$$

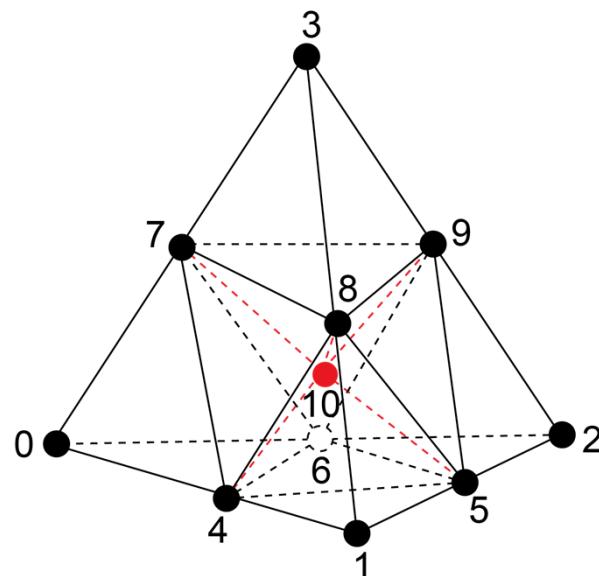
$$\bar{\mathcal{B}}_a(\boldsymbol{\xi}) = \bar{\mathcal{L}}_{a;ilk}(\boldsymbol{\xi}) \left(\frac{\partial \xi_l}{\partial X_J} \right) \mathbf{e}_i \otimes \mathbf{E}_J \otimes \mathbf{e}_k$$

$$\bar{\mathcal{L}}_a(\boldsymbol{\xi}) = \lambda_\alpha(\boldsymbol{\xi}) \delta_{ik} (M_{\alpha\beta})^{-1} \sum_{S=0}^{11} \frac{\partial N_a}{\partial \xi_l} \int_{E_S} \lambda_\beta(\boldsymbol{\xi}) \, dV_\xi \mathbf{e}_i \otimes \mathbf{a}_l \otimes \mathbf{e}_k$$

$$\bar{\mathcal{B}}_a(\boldsymbol{\xi}) = \bar{\mathcal{L}}_{a;ilk}(\boldsymbol{\xi}) \left[\bar{\mathcal{L}}_{b;JlM}(\boldsymbol{\xi}) X_{b;M} \right]^{-1} \mathbf{e}_i \otimes \mathbf{E}_J \otimes \mathbf{e}_k$$

$$\bar{B}_{aJ}(\boldsymbol{\xi}) = \bar{L}_{al}(\boldsymbol{\xi}) \left[X_{Jb} \bar{L}_{bl}(\boldsymbol{\xi}) \right]^{-1}$$

$$\bar{F}_{iJ}(\boldsymbol{\xi}) = x_{ia} \bar{B}_{aJ}(\boldsymbol{\xi})$$

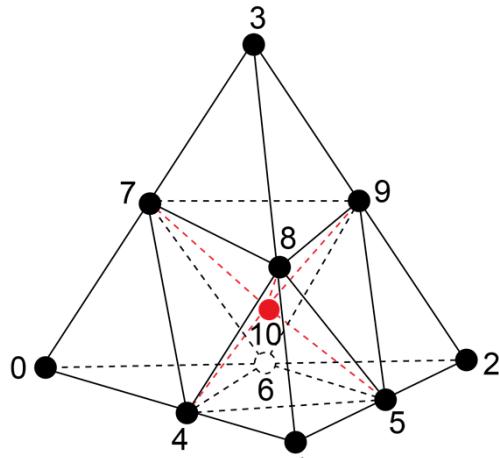


$$\bar{L}_{al}(\boldsymbol{\xi}) \equiv \bar{L}_{10 \times 3} = \frac{1}{24} \begin{pmatrix}$$

$$\begin{pmatrix} 9-60\xi_0 & 9-60\xi_0 & 9-60\xi_0 \\ -9+60\xi_1 & 0 & 0 \\ 0 & -9+60\xi_2 & 0 \\ 0 & 0 & -9+60\xi_3 \\ 70(\xi_0-\xi_1) & 2(-4-35\xi_1+5\xi_2+10\xi_3) & 2(-4-35\xi_1+10\xi_2+5\xi_3) \\ 2(-1+5\xi_1+40\xi_2-5\xi_3) & 2(-1+40\xi_1+5\xi_2-5\xi_3) & 10(\xi_0-\xi_3) \\ 2(-4+5\xi_1-35\xi_2+10\xi_3) & 70(\xi_0-\xi_2) & 2(-4+10\xi_1-35\xi_2+5\xi_3) \\ 2(-4+5\xi_1+10\xi_2-35\xi_3) & 2(-4+10\xi_1+5\xi_2-35\xi_3) & 70(\xi_0-\xi_3) \\ 2(-1+5\xi_1-5\xi_2+40\xi_3) & 10(\xi_0-\xi_2) & 2(-1+40\xi_1-5\xi_2+5\xi_3) \\ 10(\xi_0-\xi_1) & 2(-1-5\xi_1+5\xi_2+40\xi_3) & 2(-1-5\xi_1+40\xi_2+5\xi_3) \end{pmatrix}$$

This is exact. Evaluate for your integration scheme.

Volume-averaged formulation does not lock

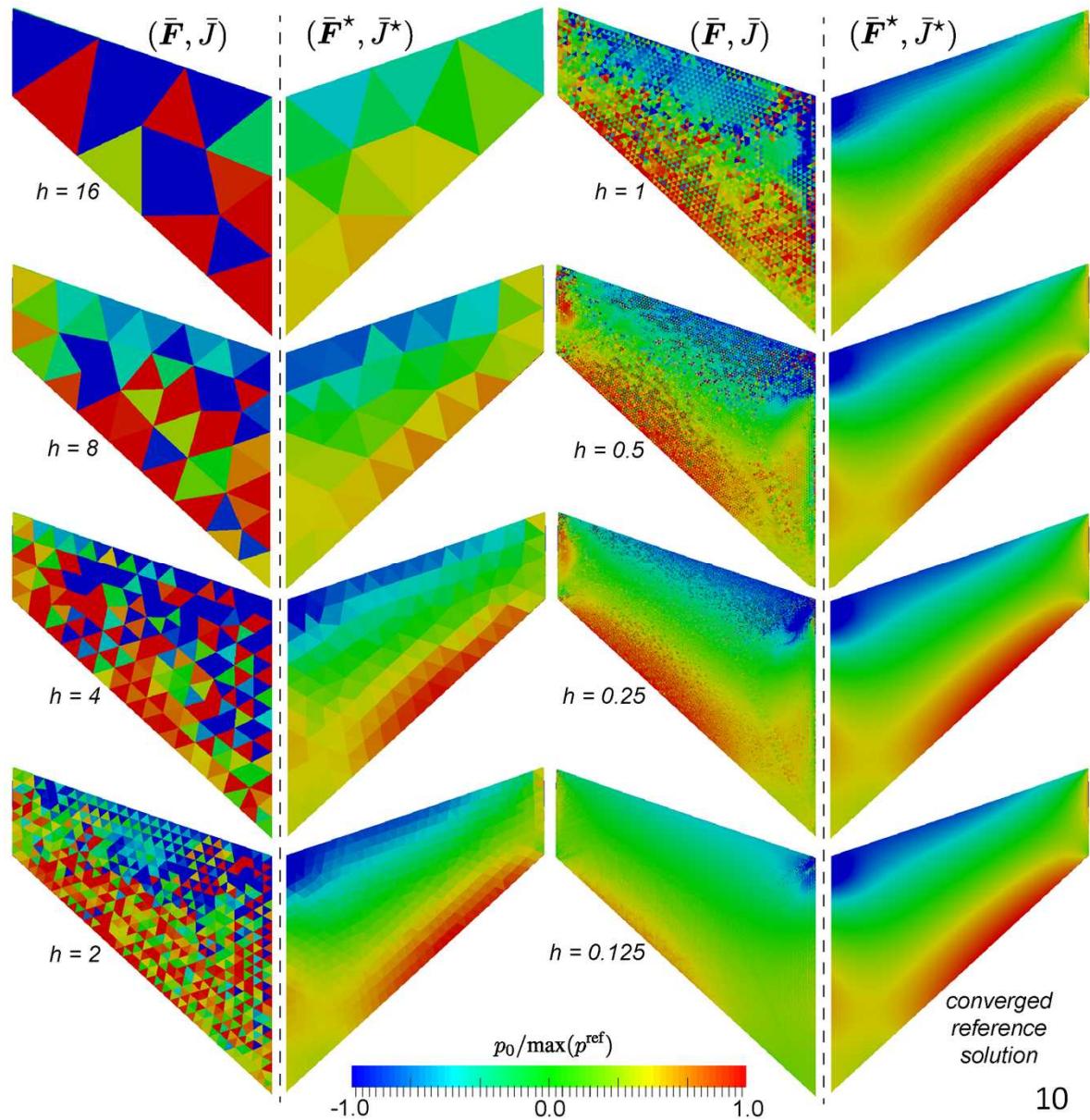


Volume-averaged formulation
does not exhibit spurious
pressure oscillations.

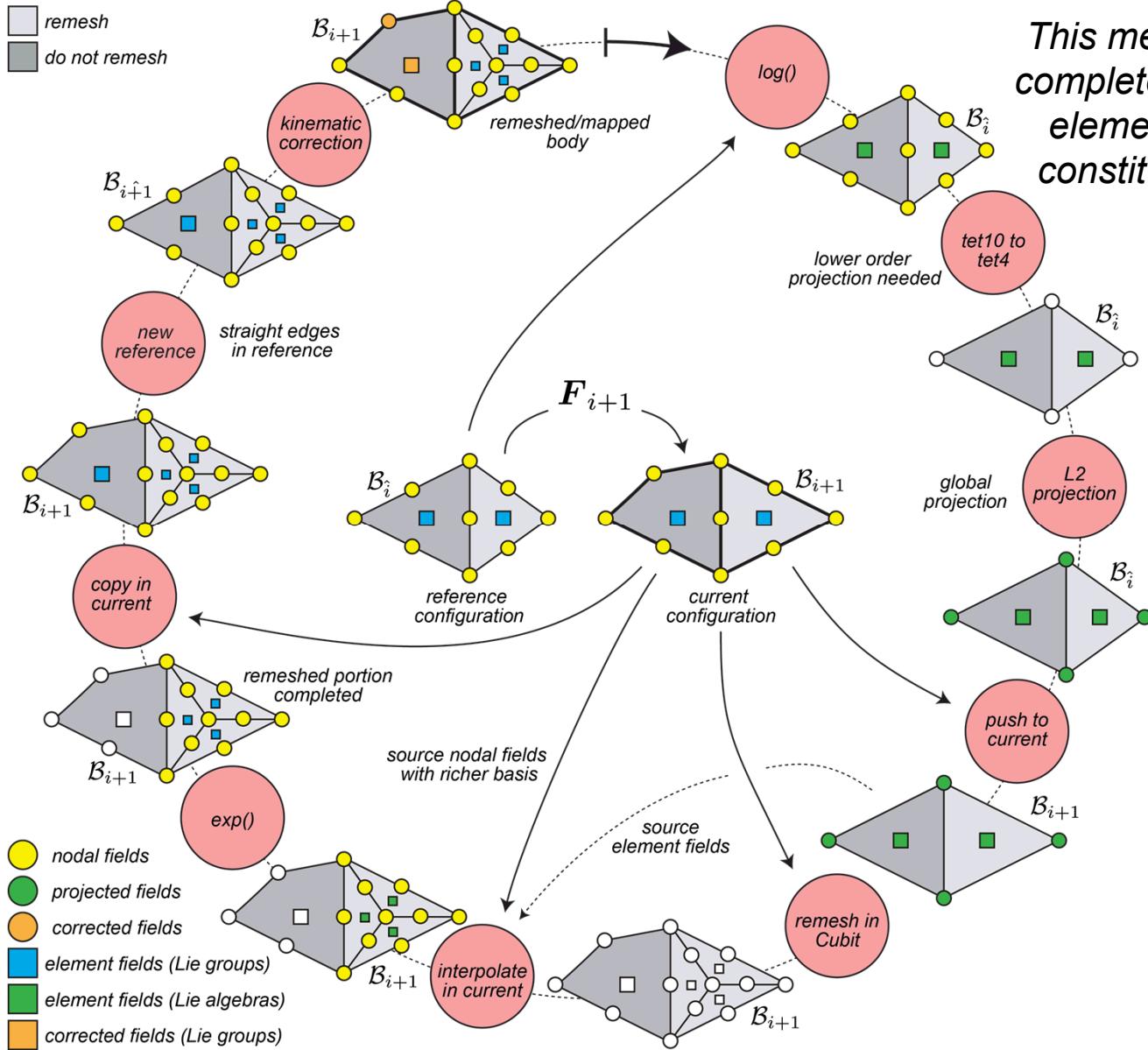
$$\bar{\mathbf{F}}^*(\xi) := \left(\frac{\bar{J}^*}{\bar{J}(\xi)} \right)^{\frac{1}{3}} \bar{\mathbf{F}}(\xi)$$

$$\bar{J}^* := \frac{\int_{\Omega} \bar{J} \, dV}{\int_{\Omega} dV}$$

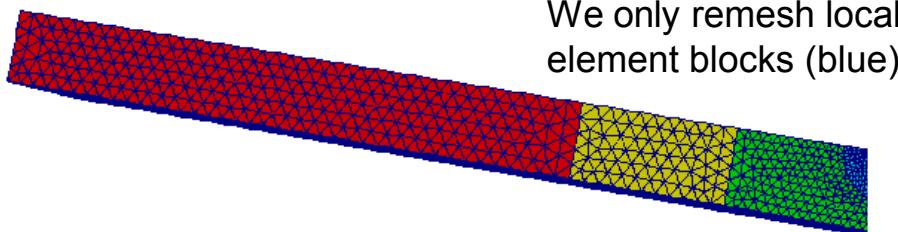
$$\bar{p}^* := \frac{1}{V_{\Omega}} \int_{\Omega} \text{tr} \frac{\partial A(\mathbf{F}^*)}{\partial \bar{\mathbf{b}}^*} \, dV$$



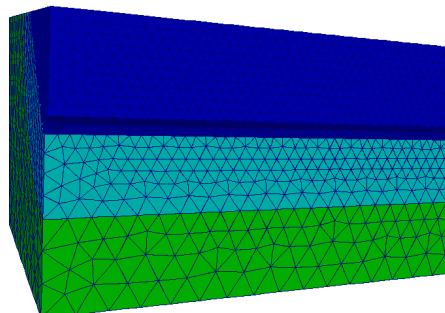
Accommodating local remeshing



Can we now model the loss of load-bearing capacity? Yes

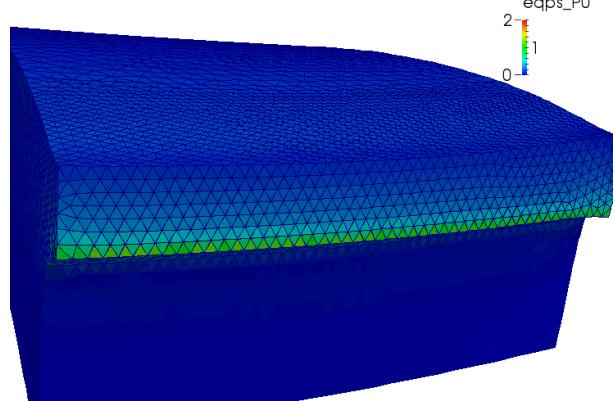
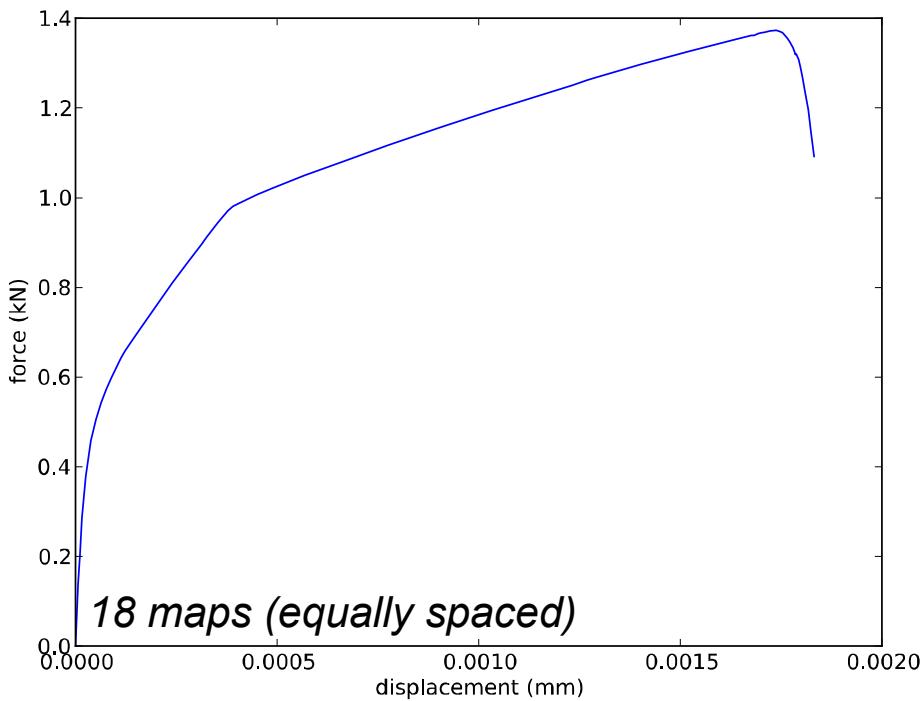


Composite-Tet10
 Elements: 110,944
 Nodes: 163,444
 Yield stress: 196 MPa
 Hardening: 2360 MPa
 Recovery: 1.3

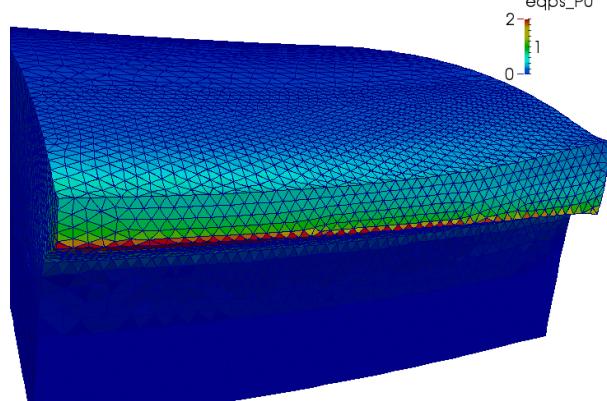


$$\sigma_y = Y + \kappa$$

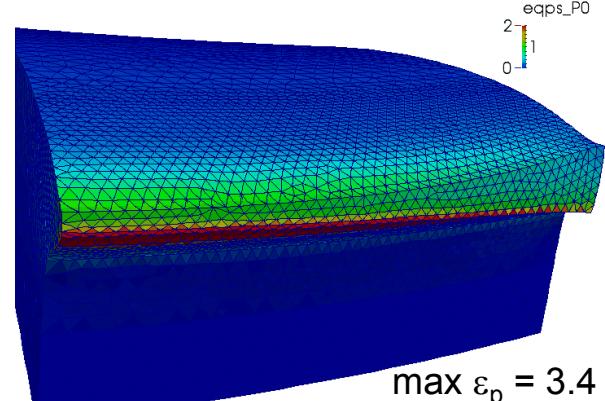
$$\dot{\kappa} = [H - R\kappa] \dot{\epsilon}_p$$



12th map



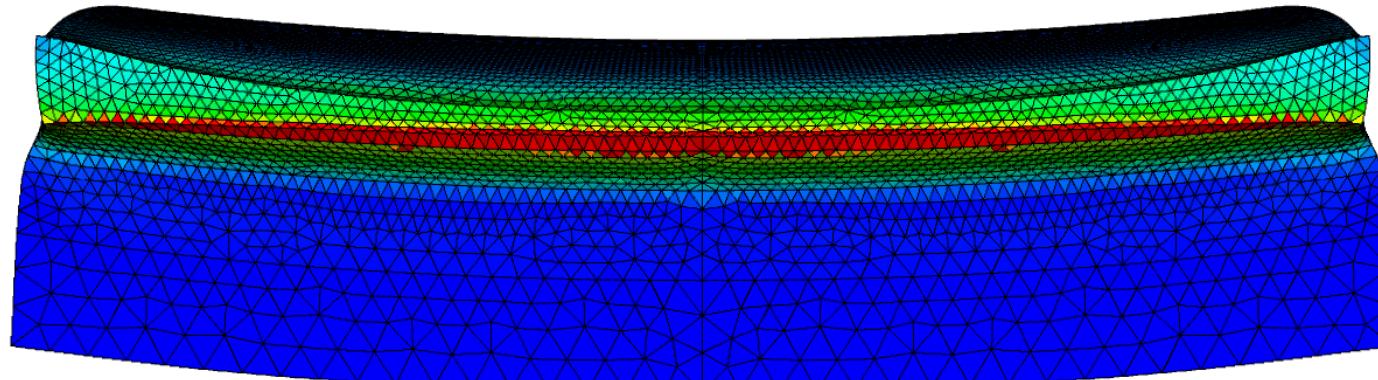
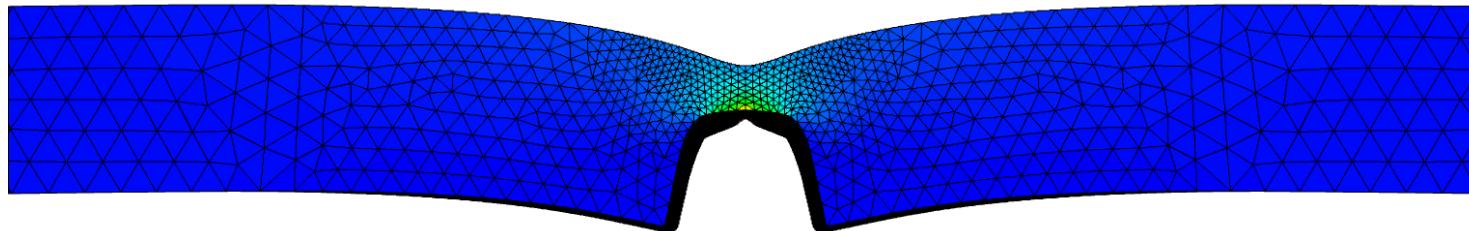
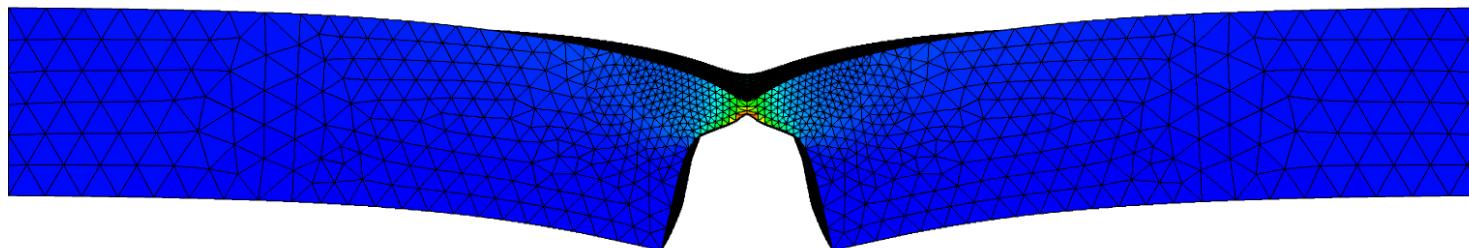
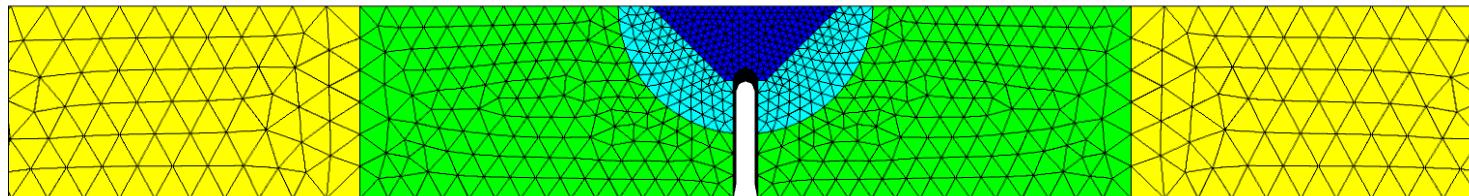
17th map



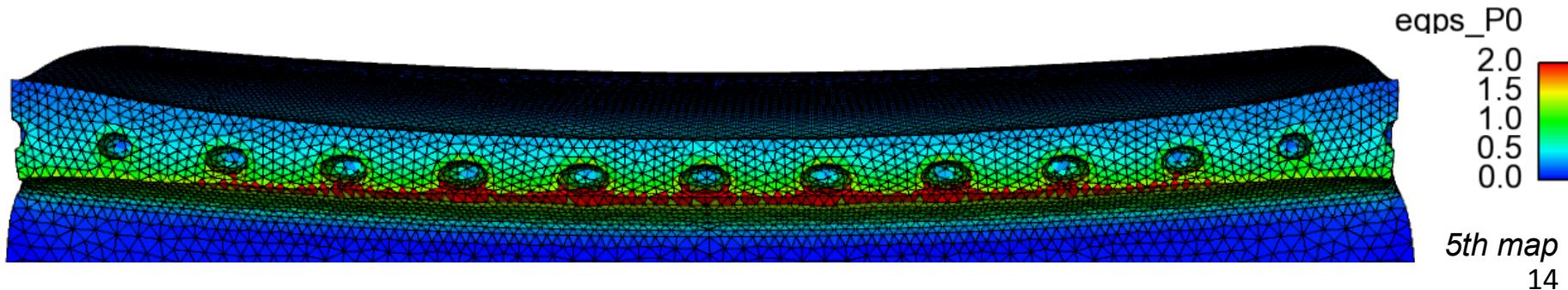
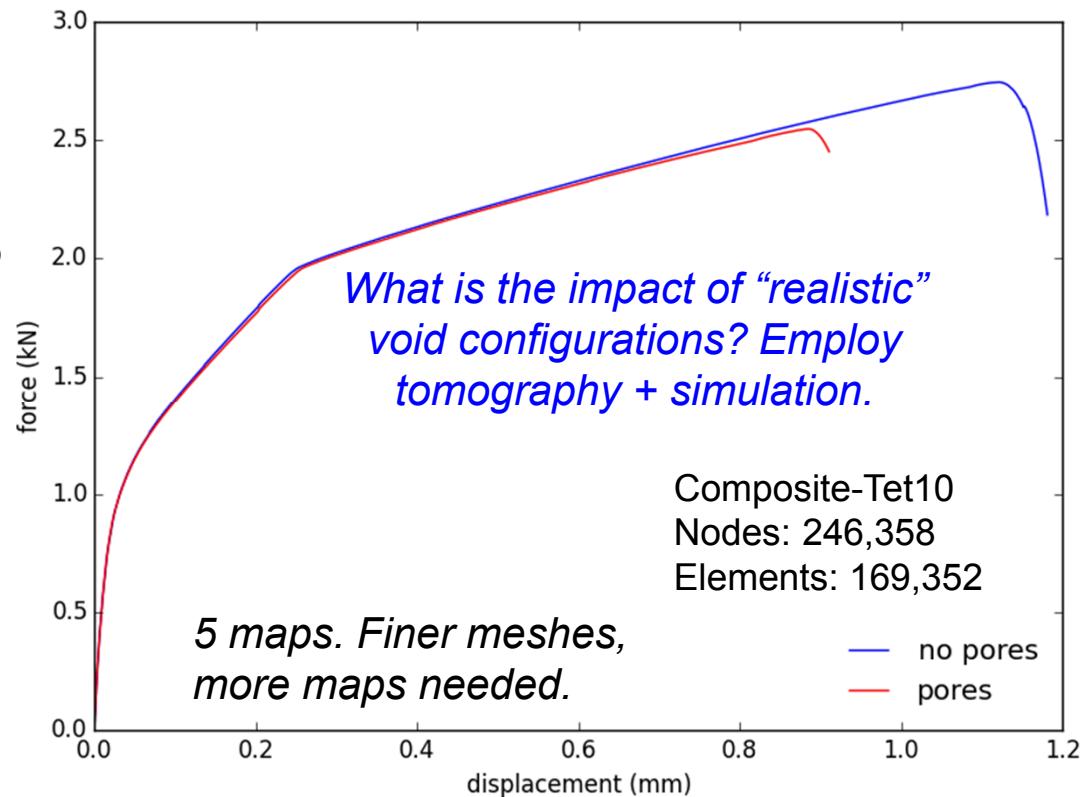
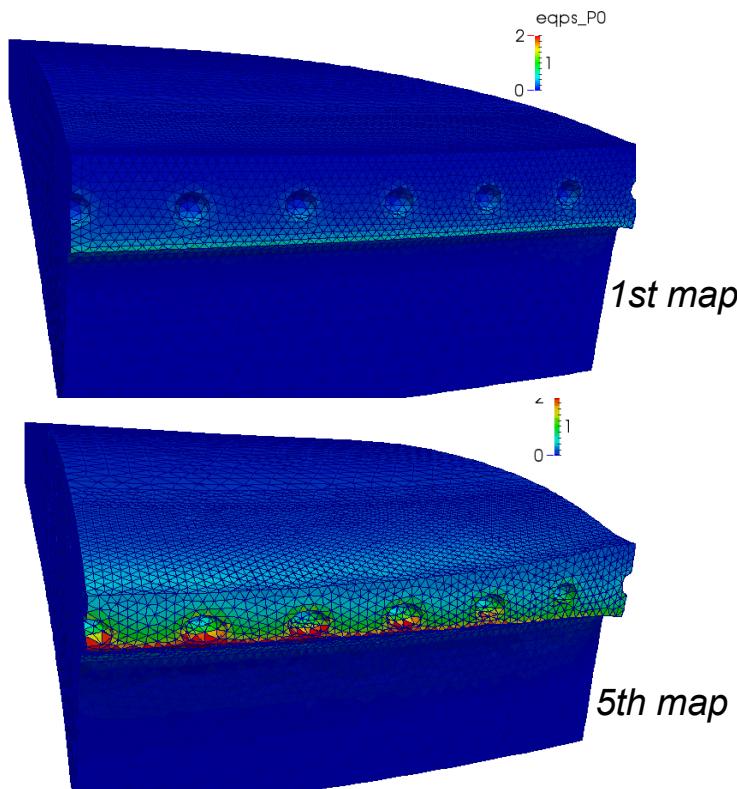
18th map

12

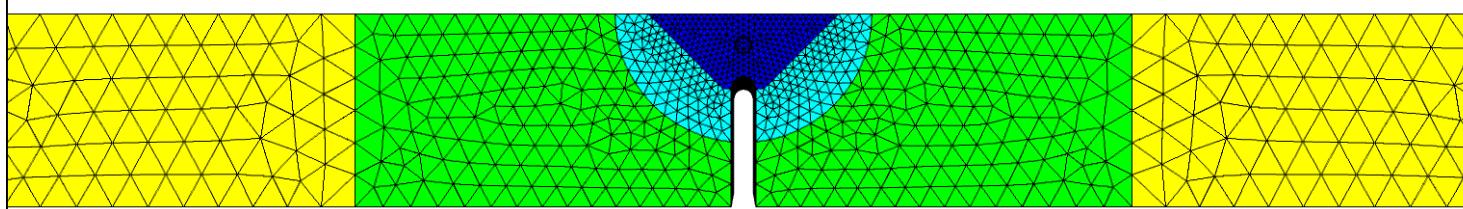
Additional interior and exterior views of necking



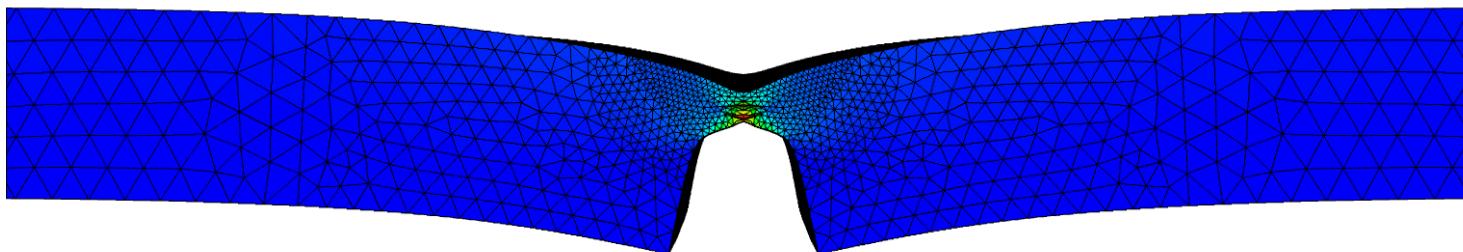
Progress in modeling the evolution of pore structures



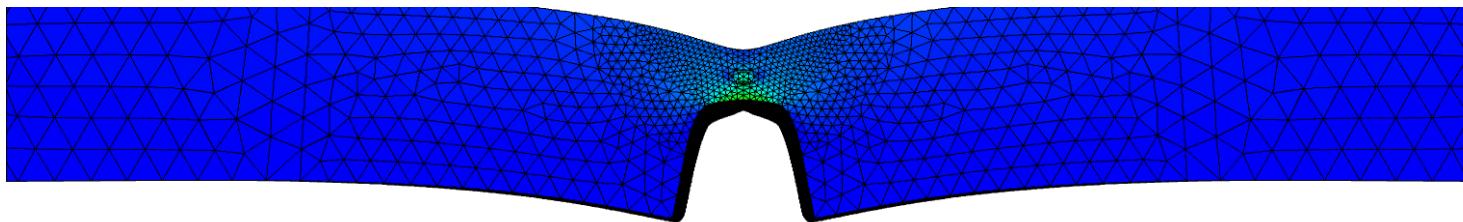
Additional interior and exterior views of necking w/pores



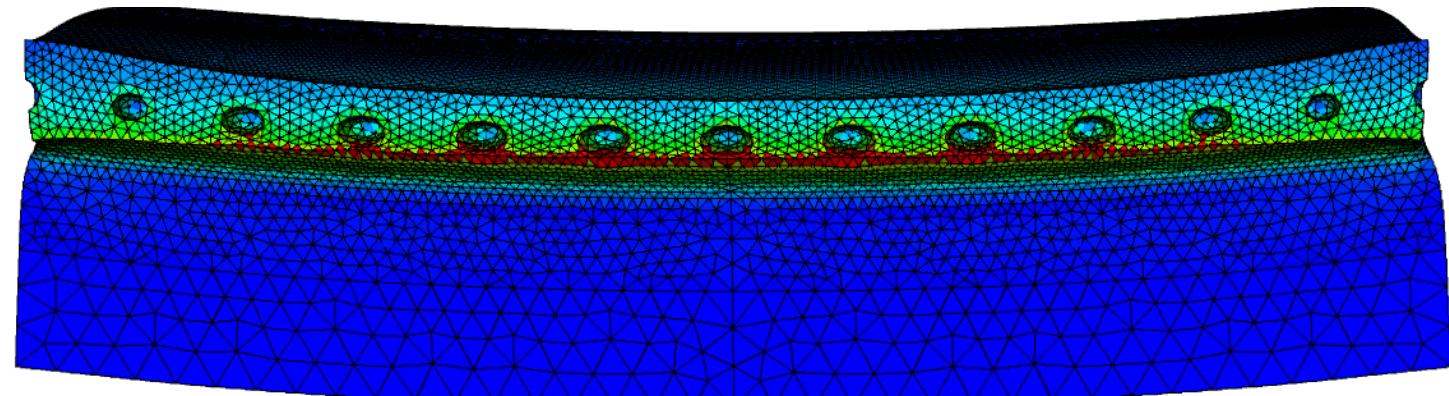
undeformed mesh
with notch



necking at
mid-plane



necking at
surface



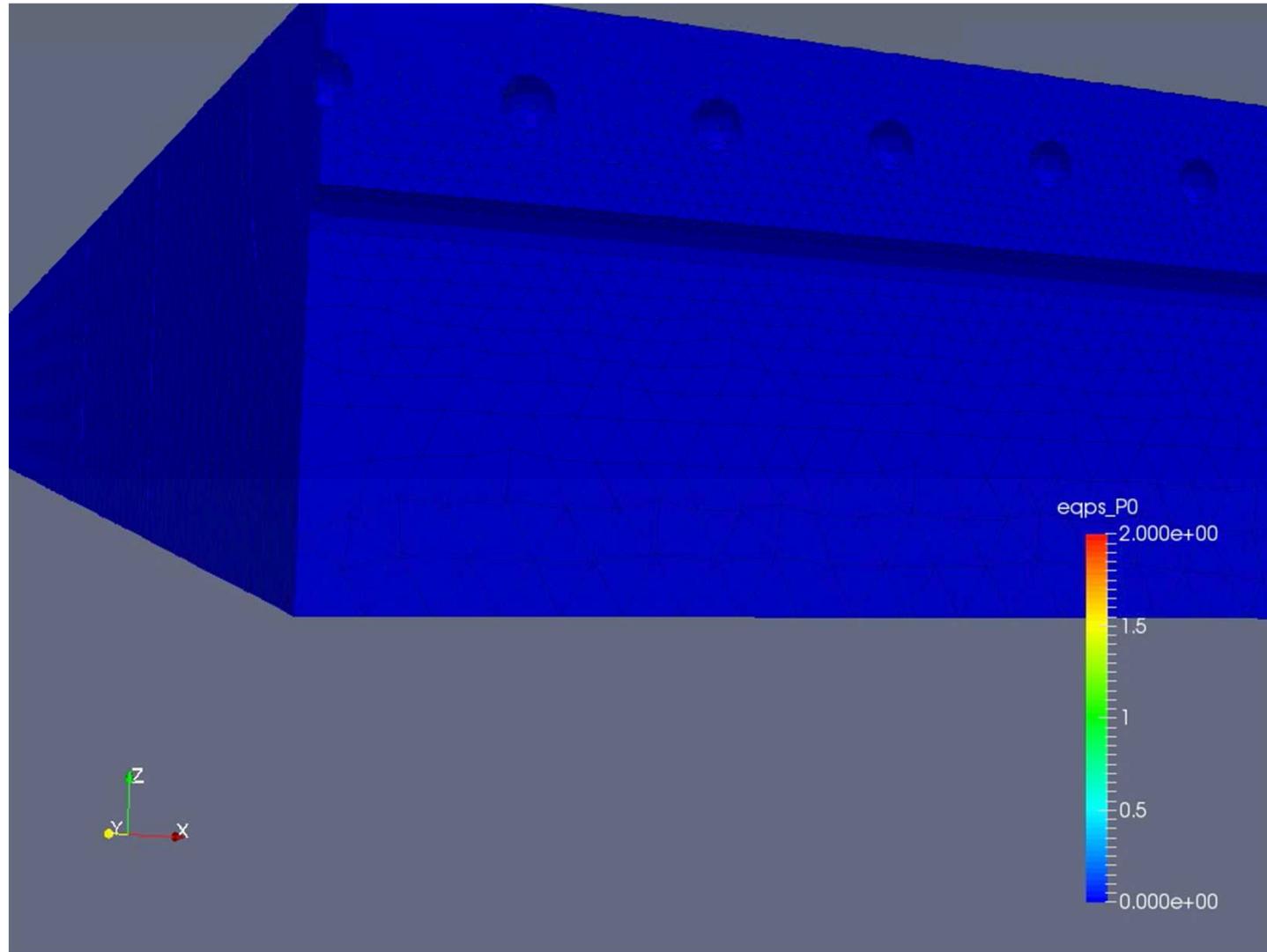
eqps_P0
2.0
1.5
1.0
0.5
0.0

$\max \varepsilon_p = 2.5$

5th map

Increasing the number of mappings

Animation illustrating the deformation process with 31 maps

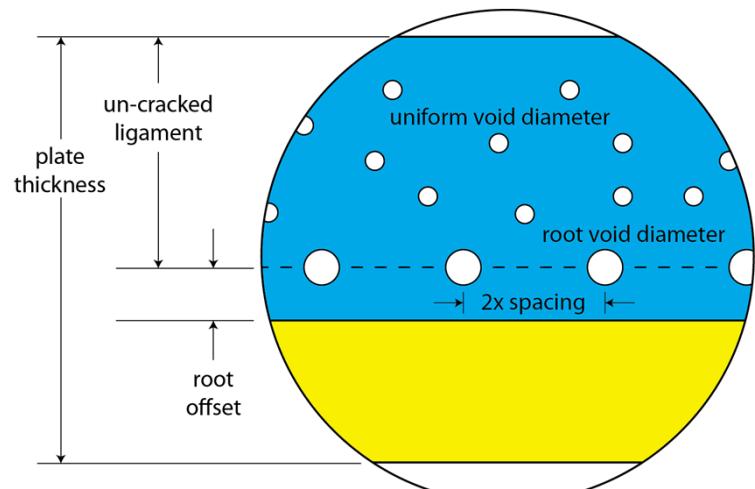
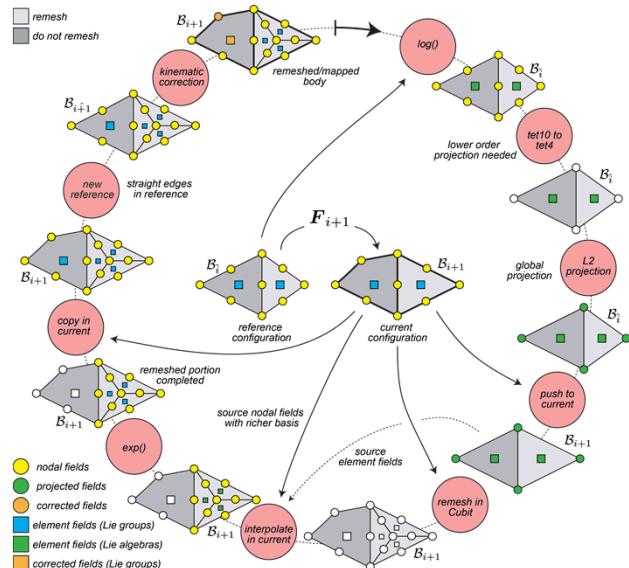


Conclusions and Path forward

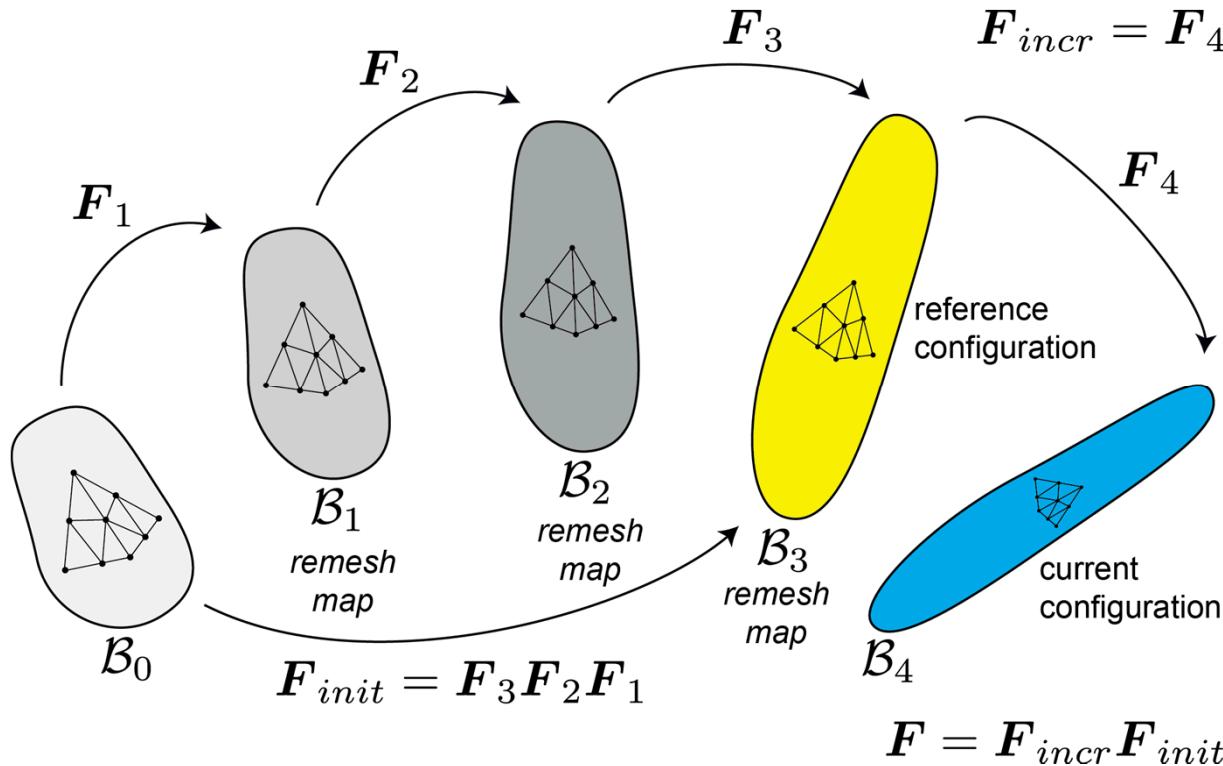
- *mapLL* ensures a sound theoretical basis
- Tetrahedral elements permit discretization*
- Composite-tetrahedral elements resolve ISVs
- New reference configuration enables solution
- We are able to predict the load-bearing capacity
- General methodology for modeling localization

Path forward

- Increase robustness of remeshing
 - Mesh geometry in Cubit problematic
- Re-examine convergence
 - Mesh refinement
 - # maps / # intervals
- Model idealized void configurations



Adopting a new reference configuration



- Prior work on hexahedral elements maintained the reference configuration
- Elements degrade in the reference configuration - T-L element integrate in reference
- We now adopt a new reference configuration and map \mathbf{F}_{init} (which lives in a Lie Group)