

# *Assessing Gaseous Hydrogen Assisted Fatigue Crack Growth Susceptibility of Pipeline Steel Weld Fusion Zones and Heat Affected Zones*

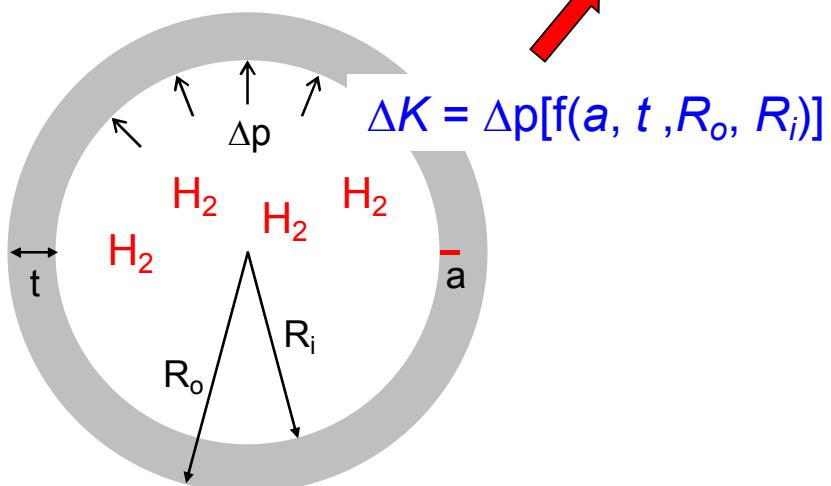
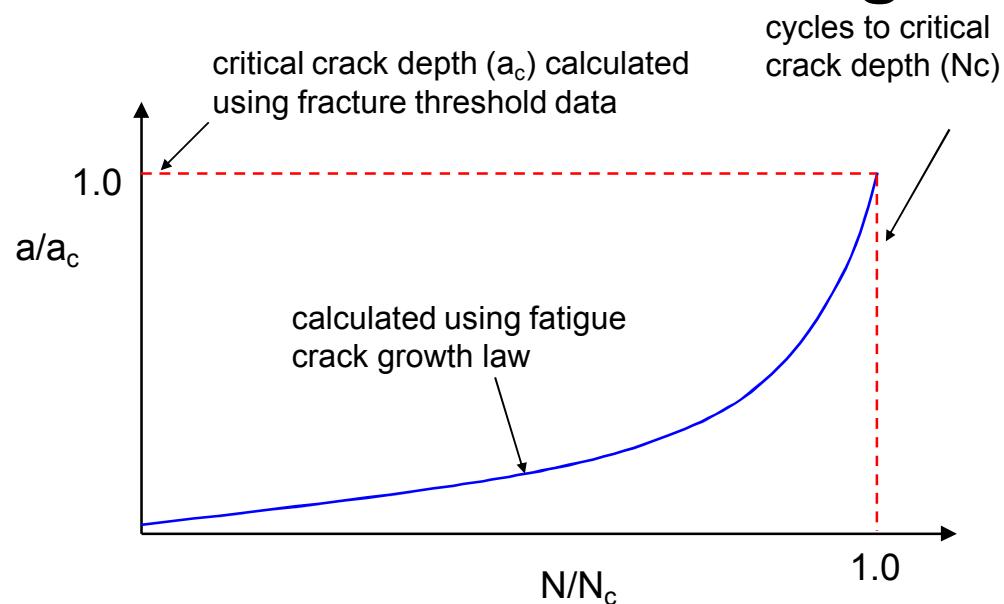
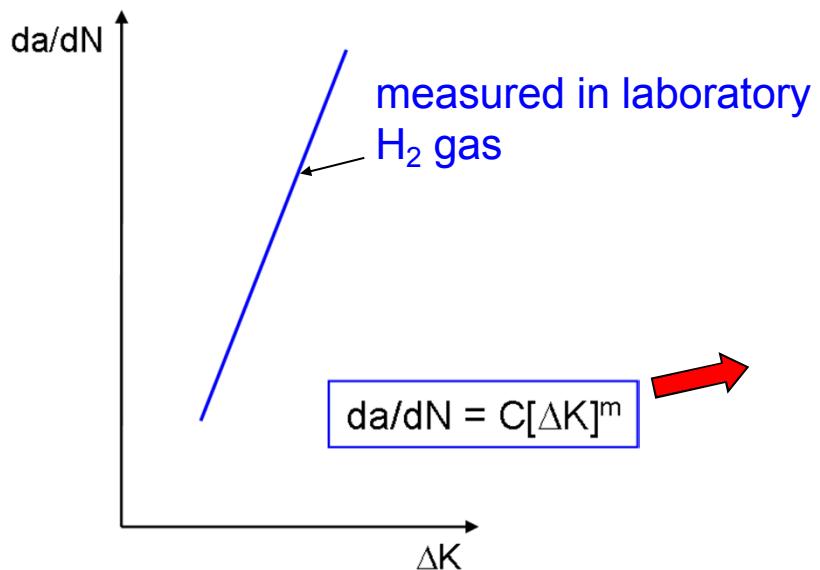
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**Sandia National Laboratories Livermore, CA**

**15<sup>th</sup> International ASTM/ESIS Symposium on Fatigue and Fracture  
Mechanics**

**Anaheim, CA**  
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# Reliability/integrity assessment framework in ASME B31.12 requires fracture data in H<sub>2</sub> gas



- Two fracture properties in H<sub>2</sub> needed
  - Fatigue crack growth law
  - Fracture threshold
- Reliability/assessment framework accommodates H<sub>2</sub> embrittlement

# Material Tested: API 5L X65 steel with GMAW

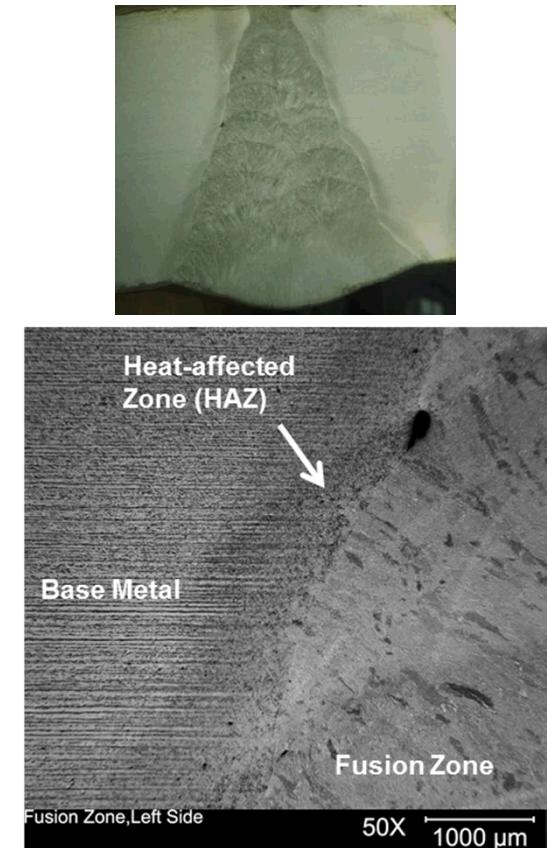
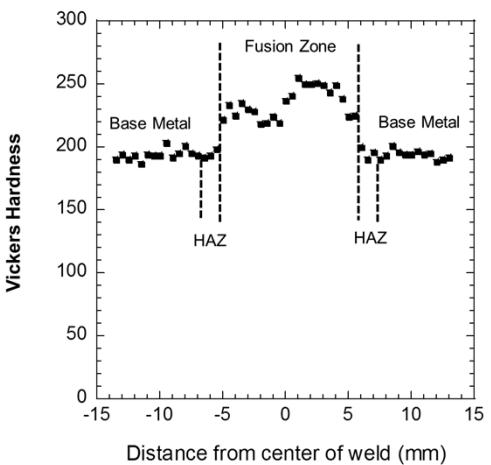
Gas Metal Arc Weld (GMAW)



508 mm OD / 25.4 mm thickness

Filler metal: Thyssen TS-6 ER70S-G

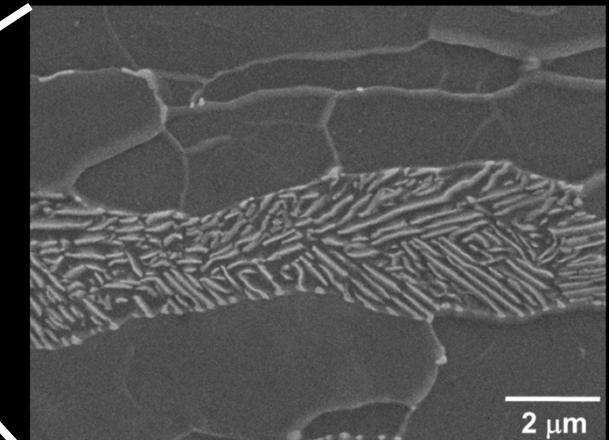
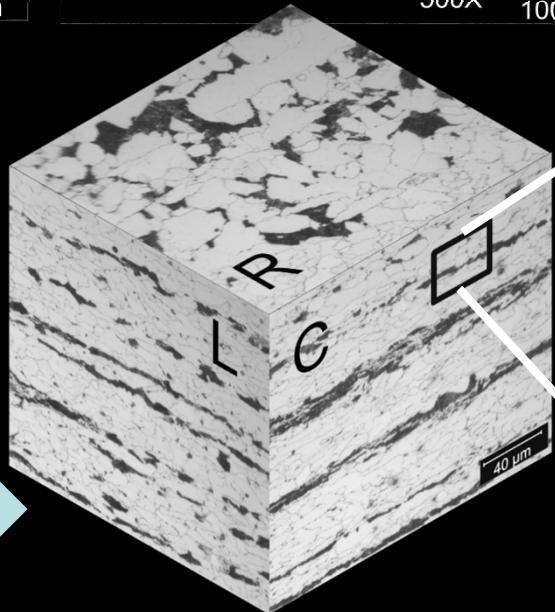
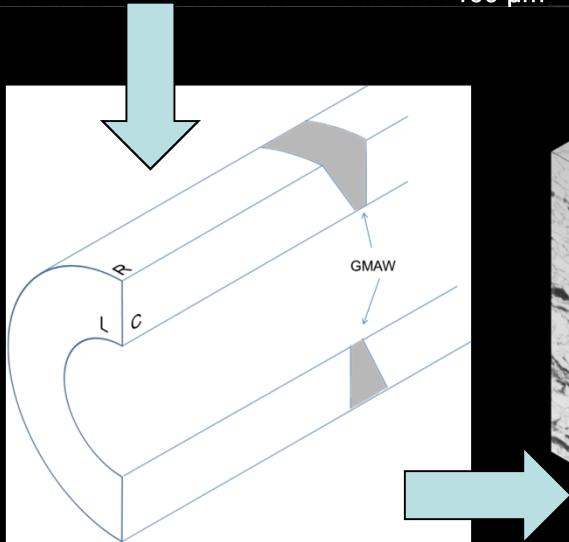
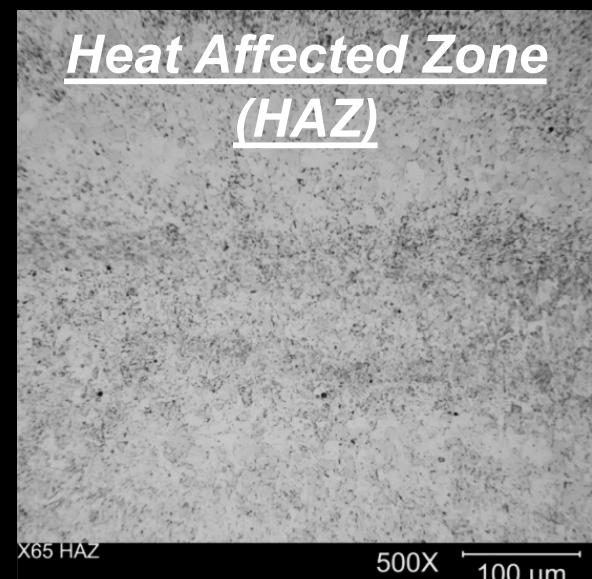
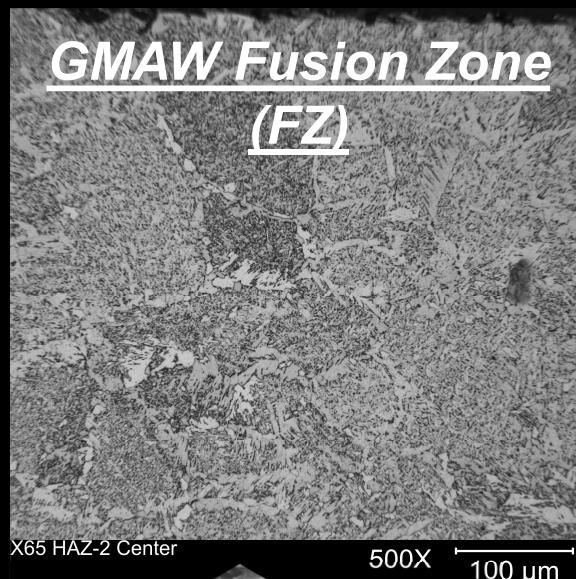
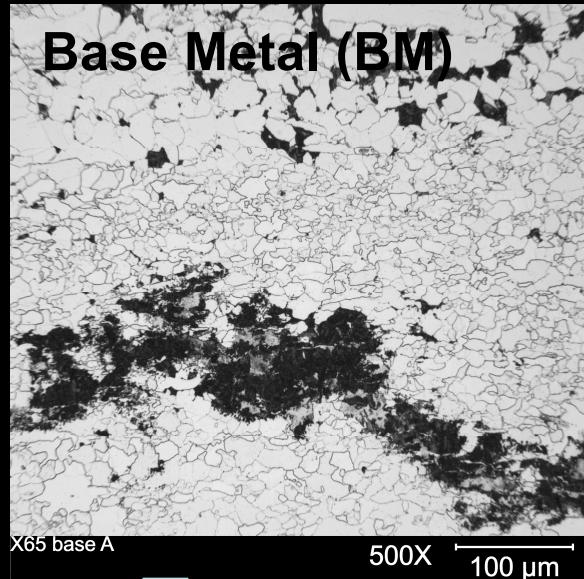
Material	YS (MPa)	UTS (MPa)
Base Material	478	564
GMAW	591	662



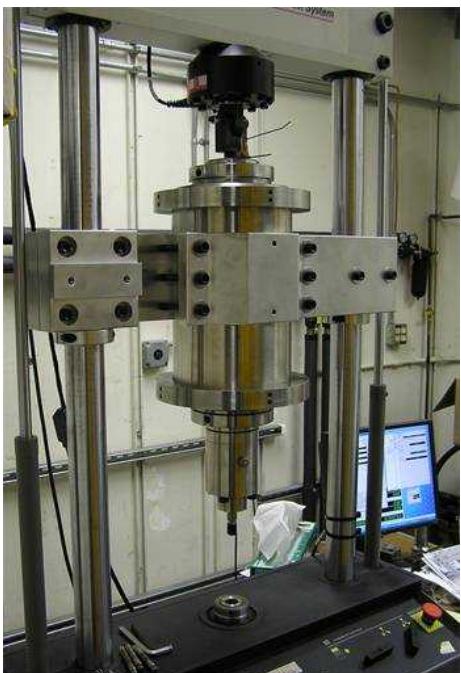
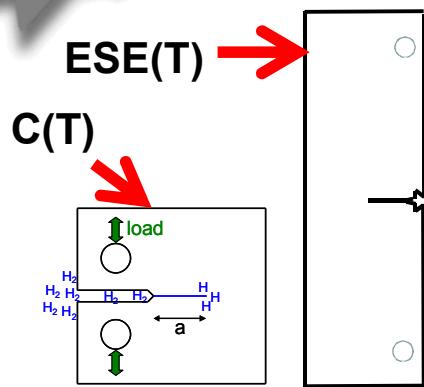
Base Metal Chemical Composition (wt %)

C	Mn	P	S	B	Si	Cu	Ni	Nb	Ti
0.08	1.53	0.01	0.001	0.002	0.32	0.024	0.038	0.039	0.002

# Microstructure of X65



# Measure fatigue crack growth in high-pressure H<sub>2</sub>



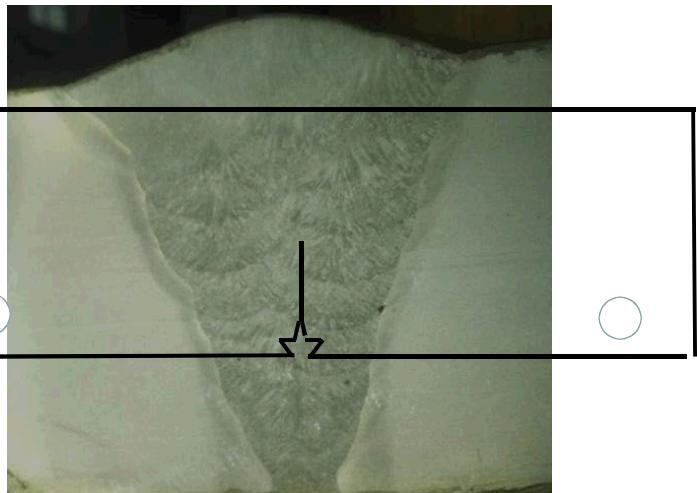
- Specimen Geometry
  - 12.7 or 6.4 mm Compact Tension, C(T)
  - 4.7 mm thick Eccentrically loaded single edge notched tension, ESE(T)
- Instrumentation
  - Internal load cell in feedback loop
  - Crack-opening displacement measured internally using LVDT or clip gage
  - Crack length calculated from compliance
- Mechanical loading
  - Triangular load-cycle waveform
  - Constant load amplitude (increasing  $\Delta K$ )
    - $R = \frac{P_{min}}{P_{max}} = 0.5$
- Environment
  - Primary supply gas: 99.9999% H<sub>2</sub>
  - Pressure = 21 MPa (3,000 psi)
  - Room temperature (295 K)



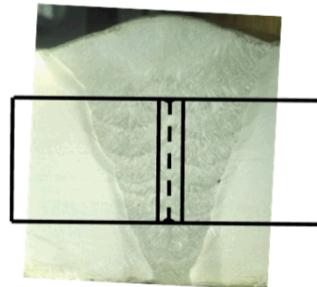
# Non-uniform crack fronts in weld FZ and HAZ specimens

Compact Tension C(T) specimens extracted from pipe

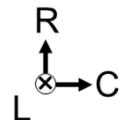
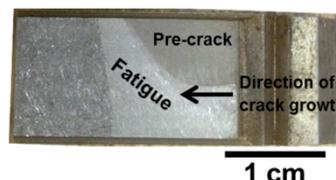
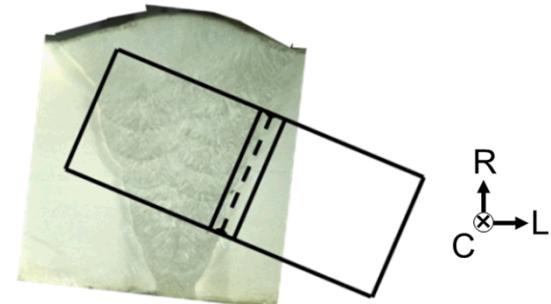
- Entire crack plane contained in FZ or HAZ
- Non-uniform crack fronts observed
- Attributed to residual stress



FZ

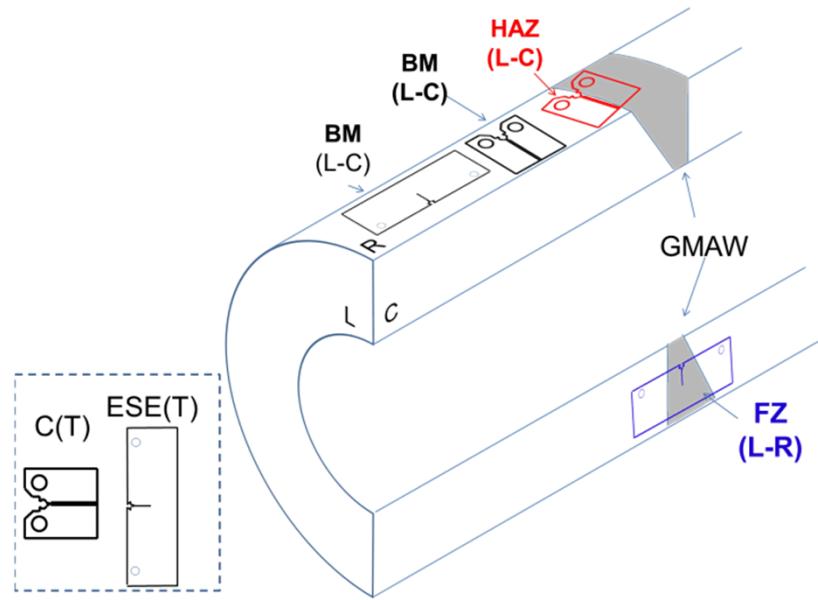
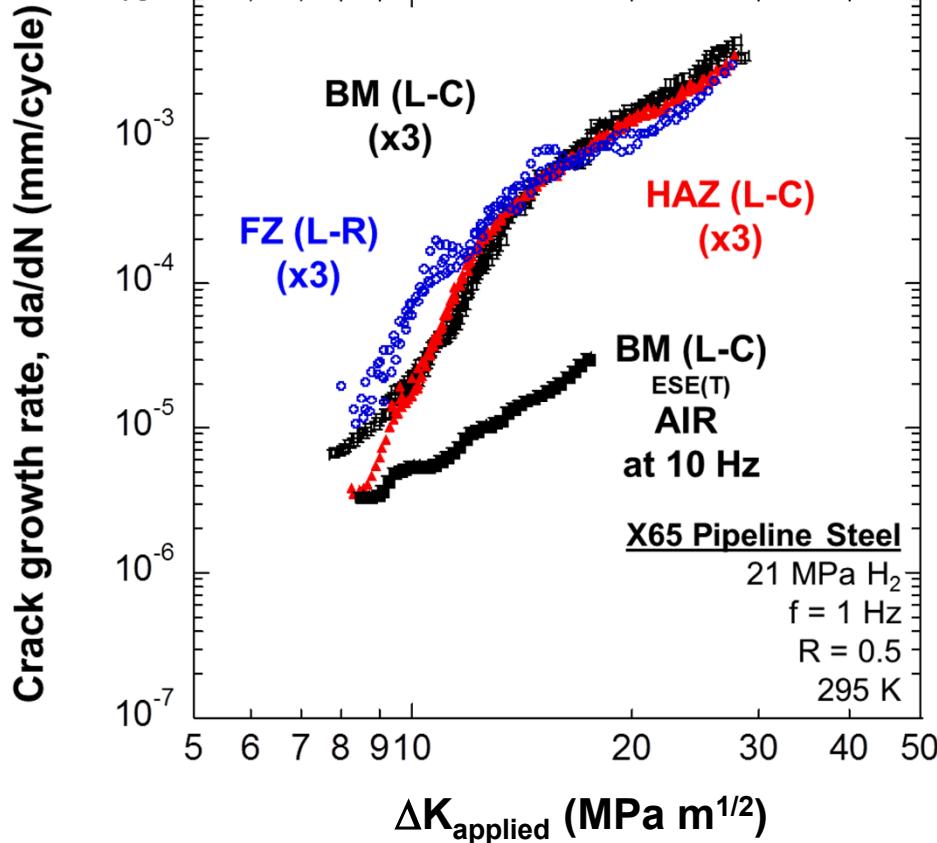


HAZ



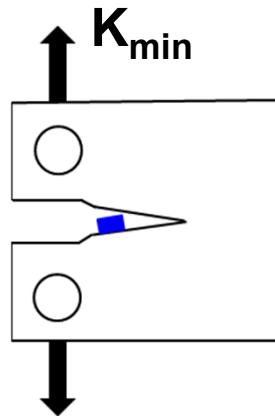
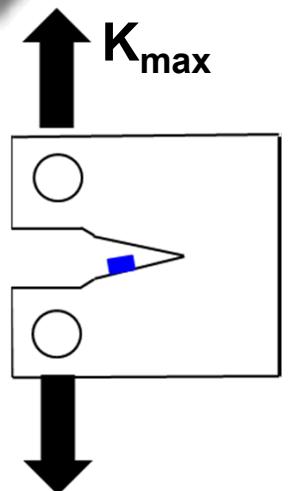
**Alternate standardized test specimen geometry necessary for weld FZ testing: ESE(T)**

# Repeatable da/dN vs. $\Delta K$ curves for BM, FZ, and HAZ in 21 MPa $H_2$

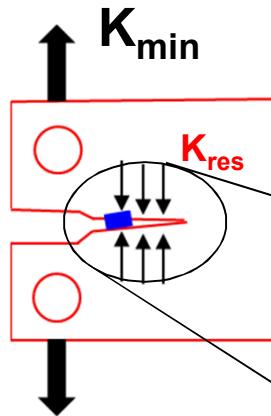
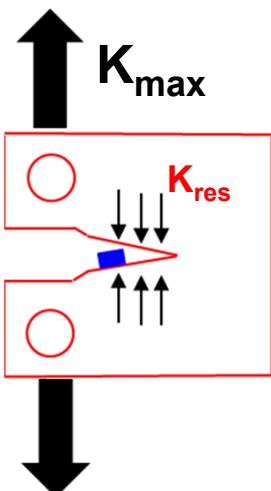


- Triplicate tests revealed repeatable results
- Results do not account for residual stress

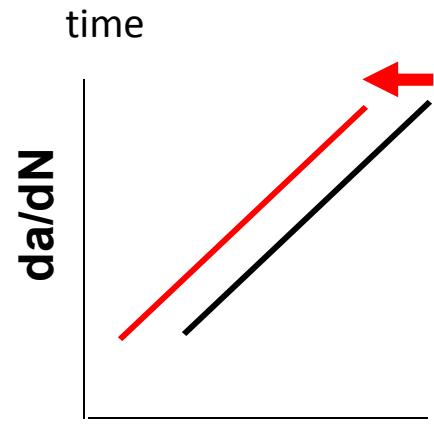
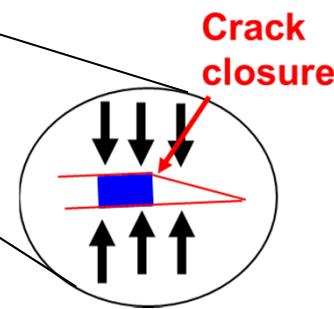
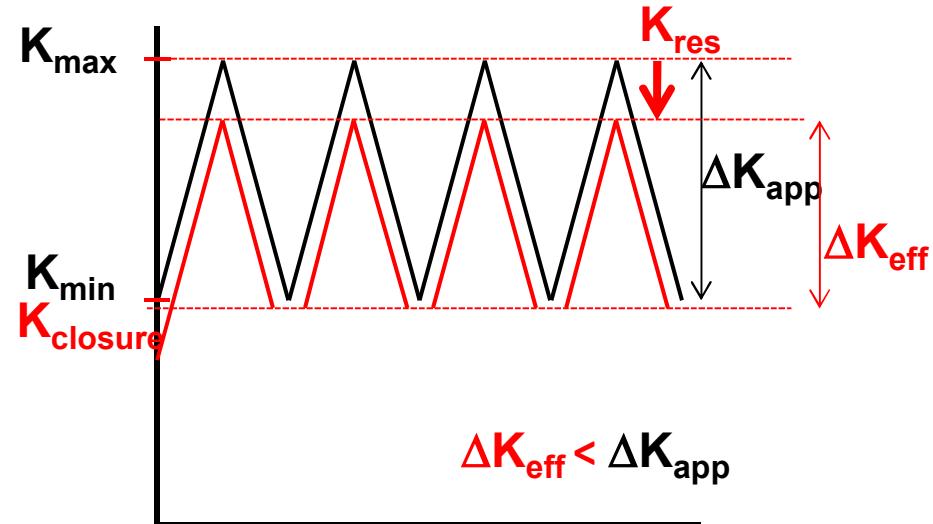
# Analysis performed to account for effect of residual stress on crack-driving force



No residual stress



Compressive residual stress



$\Delta K_{\text{eff}}$ ,  $\Delta K$

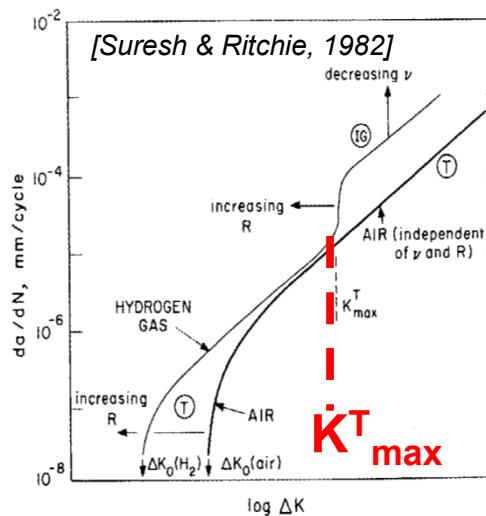
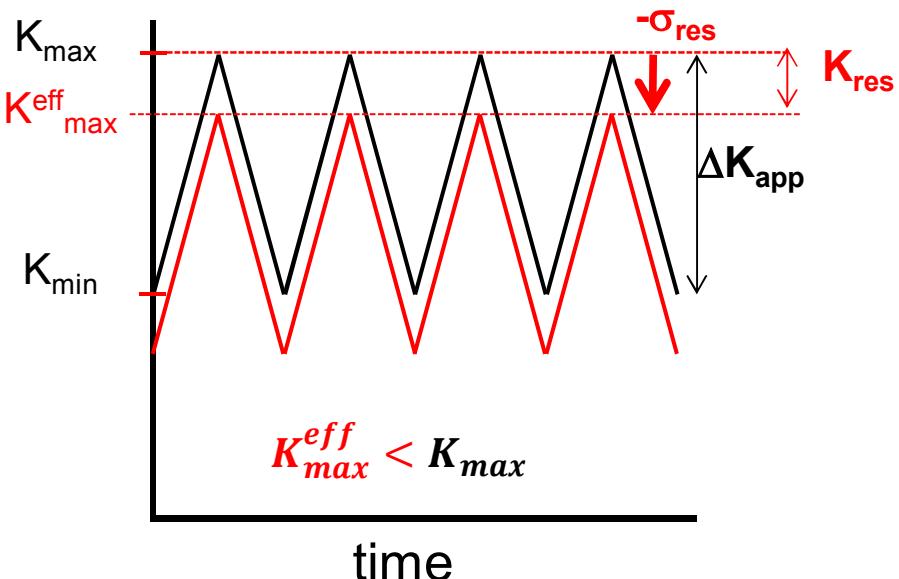


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- Compressive residual stress induces “crack closure”, reducing crack-driving force from  $\Delta K$  to  $\Delta K_{\text{eff}}$

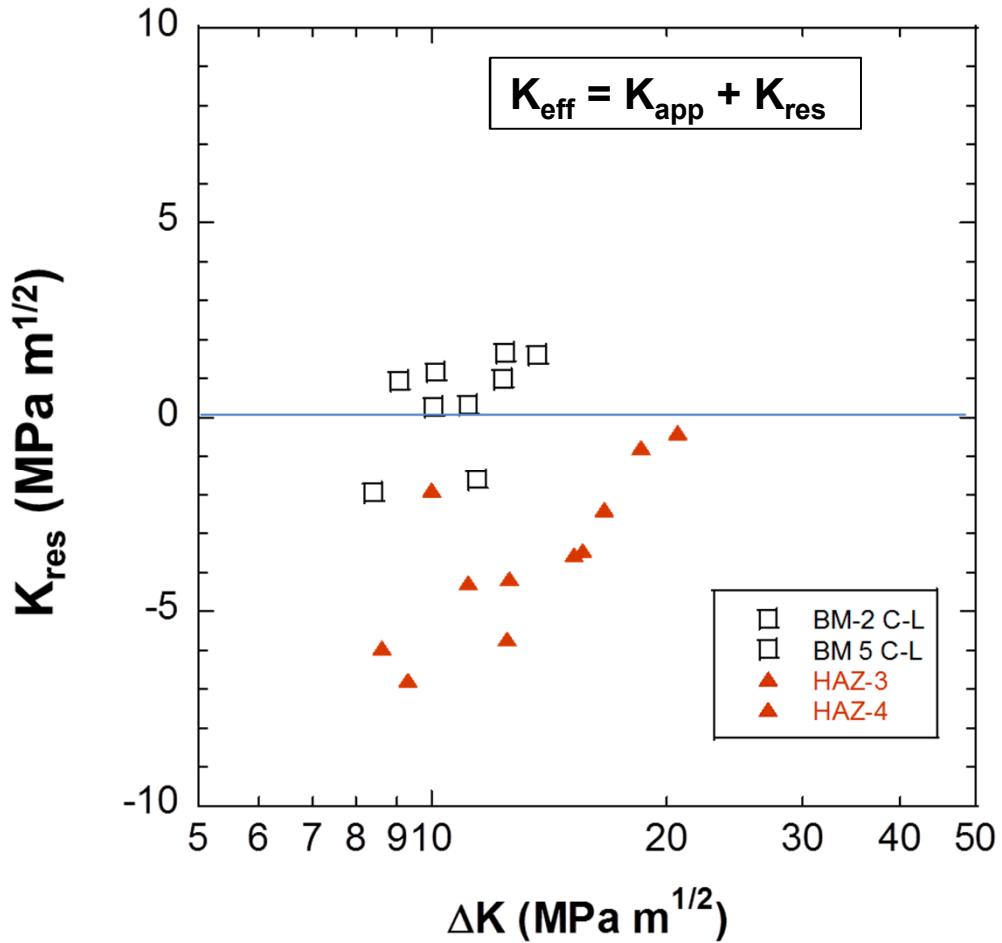
# Residual stress can reduce $K_{\max}$ ~ affect onset of HA-FCG

- Compressive residual stress will reduce  $K_{\max}$  to  $K_{\max}^{\text{eff}}$



# Compressive residual stress can reduce $K_{\text{eff max}}$

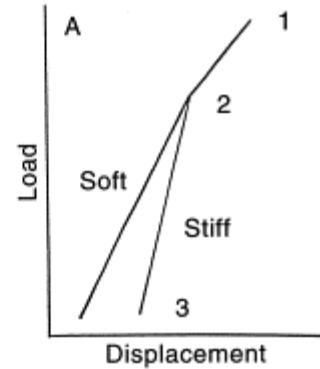
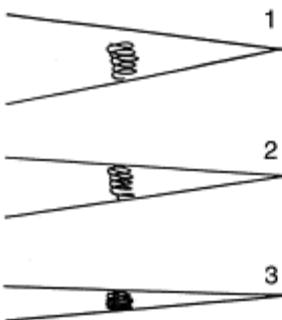
- Calculated residual stress effect on  $K_{\text{max}}$  →  $K_{\text{res}}$
- BM exhibited negligible  $K_{\text{res}}$
- HAZ exhibited greater  $K_{\text{res}}$



(Lados, 2007)

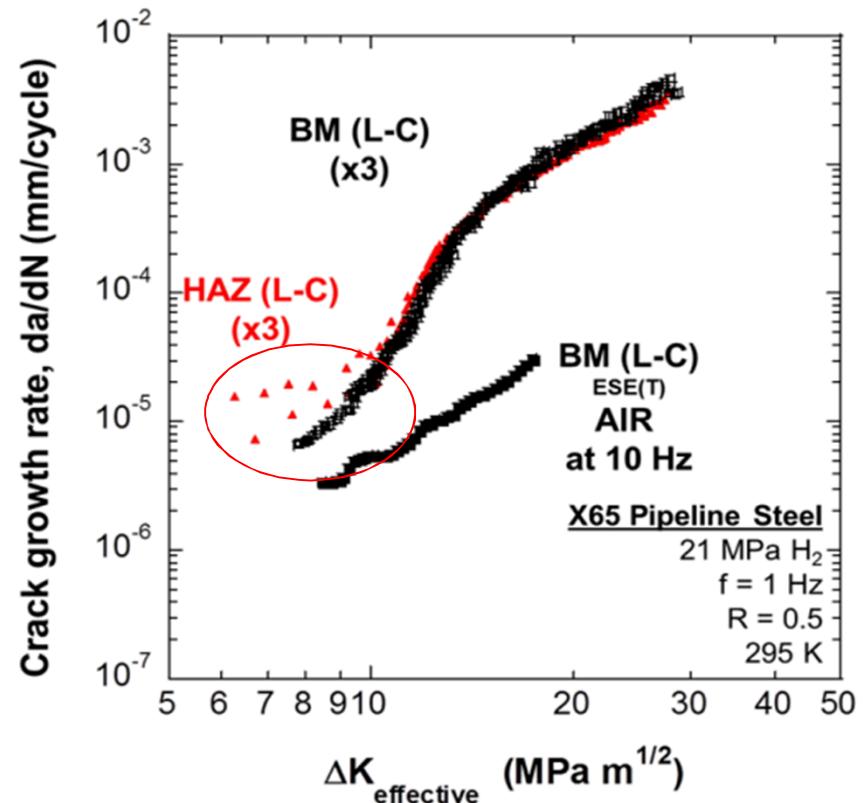
Reduction in  $K_{\text{eff max}}$  could artificially shift the onset of HA-FCG to higher  $\Delta K$  values

# Crack closure reduces $\Delta K_{\text{effective}}$



(Donald, 1997)

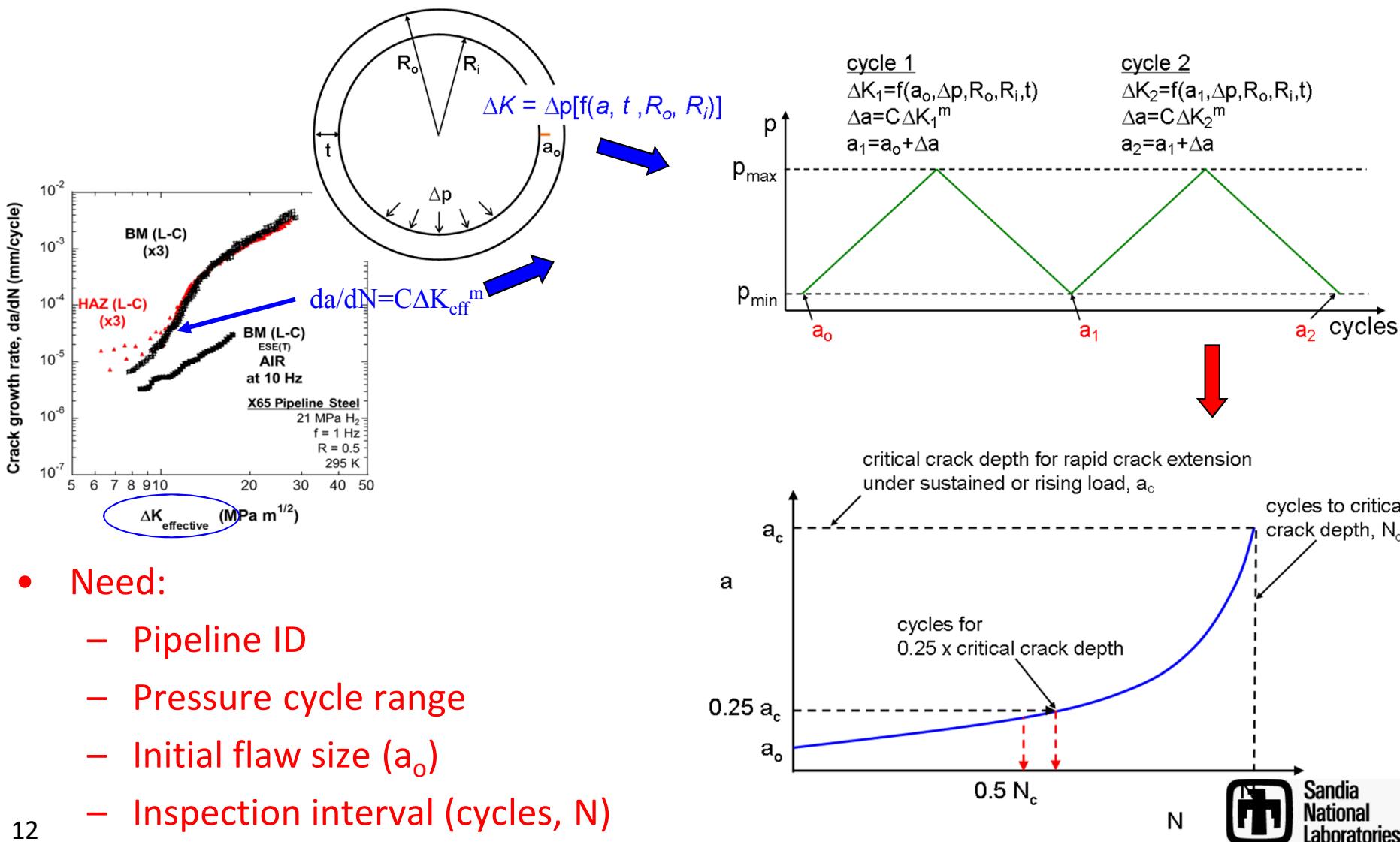
- Crack closure: crack remains closed for portion of loading cycle
- Adjusted Compliance Ratio (ACR) method implemented to remove effects of closure  
→  $\Delta K_{\text{effective}}$

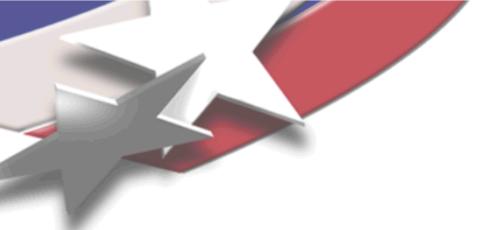


- Reveals higher FCGR in HAZ compared to BM

Highlights importance in comparing intrinsic fatigue behavior

# Measured fatigue crack growth relationships can be used to specify wall thickness for H<sub>2</sub> pipelines





# Summary

- Completed triplicate fatigue measurements of X65 pipe girth weld, HAZ, and BM in 21 MPa H<sub>2</sub> gas, results were reproducible
  - Welds appear to be more susceptible to H<sub>2</sub>-accelerated fatigue crack growth compared to base metal
  - HAZ results show slightly higher FCGR when extrinsic factors such as crack closure and residual stress effects were removed
- Comparison of intrinsic FCG behavior (e.g.  $\Delta K_{\text{eff}}$ ) is necessary for evaluating microstructure performance in hydrogen
  - Requires removal of extrinsic effects specific to pipe (crack closure, residual stress, etc...)



# Acknowledgements / References

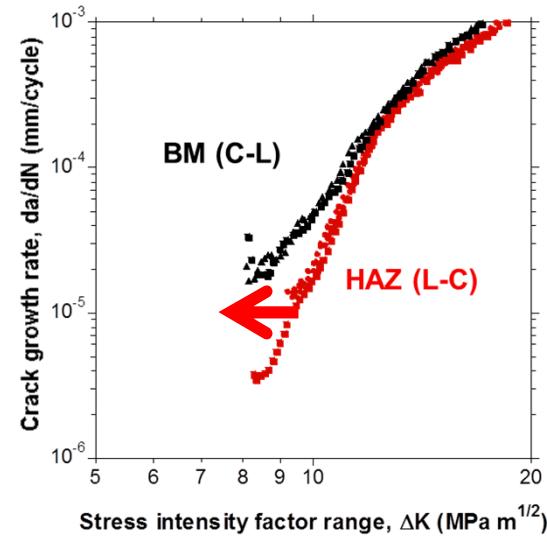
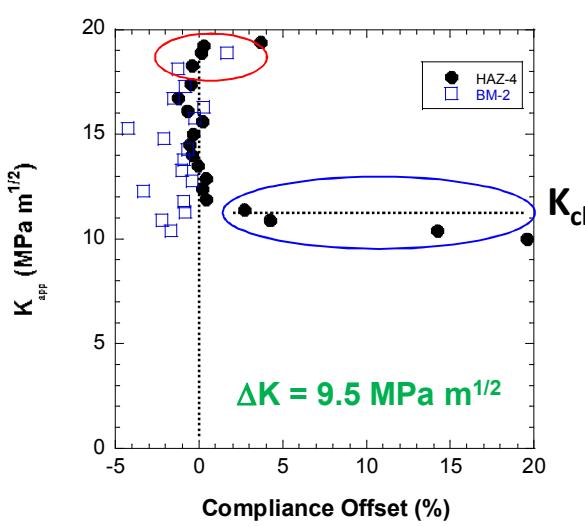
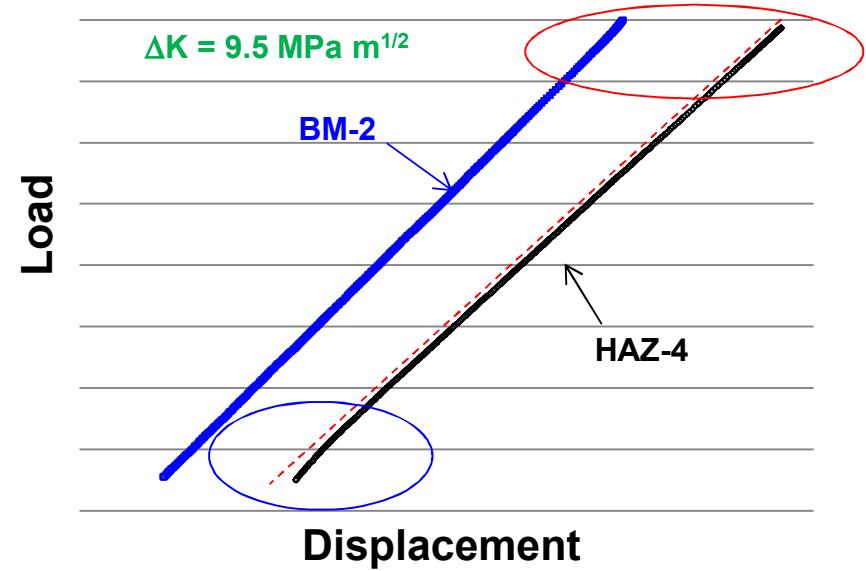
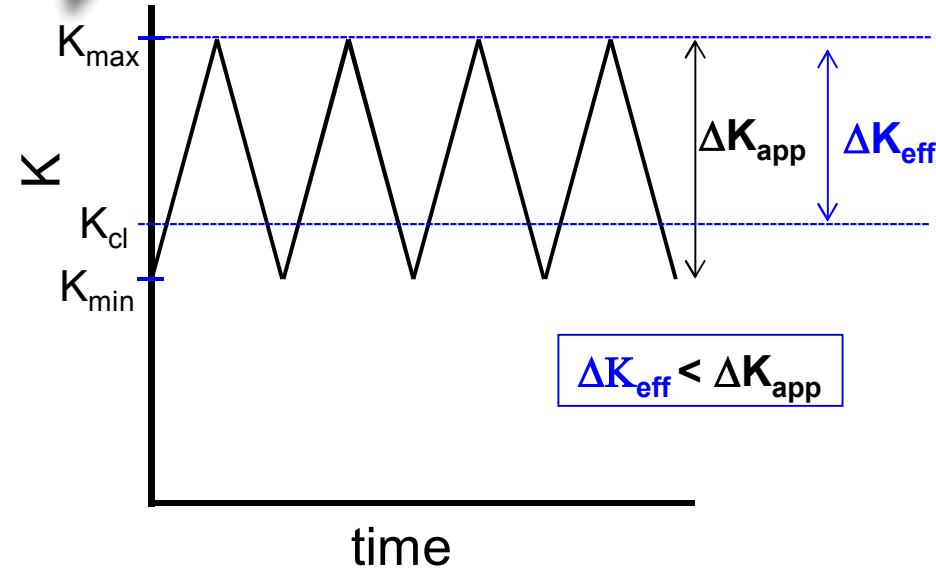
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- [1] Neeraj, T., Gnaupel-Herold, T., Prask, H.J., Ayer, R., "Residual stresses in girth welds of carbon steel pipes: neutron diffraction analysis," *Science and Technology of Welding and Joining*. Vol. 16, No.3, pp. 249-253, 2011.
- [2] Suresh, S. and Ritchie, R.O. "Mechanistic dissimilarities between environmentally influenced fatigue-crack propagation at near-threshold and higher growth rates in lower strength steels," *Metal Science*, vol. 16, 1982, pp. 529-538.
- [3] Cialone, H.J. and Holbrook, J.H. "Effects of gaseous hydrogen on fatigue crack growth in pipeline steel." *Metallurgical Transactions A*, vol. 16A, 1985, pp. 115-122.
- [4] Cialone, H.J. and Holbrook, J.H. "Microstructure and fractographic features of hydrogen-accelerated fatigue crack growth in steels" in *Microstructural Science* Vol. 14, Eds. M.R. Louthan, I. LeMay, and G.F. Vander Voort, ASM, Metals Park, OH, (1987) pp. 407-422.
- [5] Kemp, P.M.J., "Fatigue Crack Closure- A Review." Technical Report TR 90046, Sept. 1990. Royal Aerospace Establishment.
- [6] Donald, J.K., "Introducing the Compliance Ratio Concept for determining effective stress intensity," *International Journal of Fatigue*, vol. 19, Issue 93, 1997, pp. 191-195.
- [7] Lados, D., Apelian, D., Donald, J., "Fracture mechanics analysis for residual stress and crack closure corrections," *International Journal of Fatigue*, vol. 29, 2007, pp. 687-394.



# Back-up Slides

# Crack closure appears to occur in HAZ even at $R = 0.5$ , resulting in $\Delta K_{\text{eff}} < \Delta K_{\text{app}}$



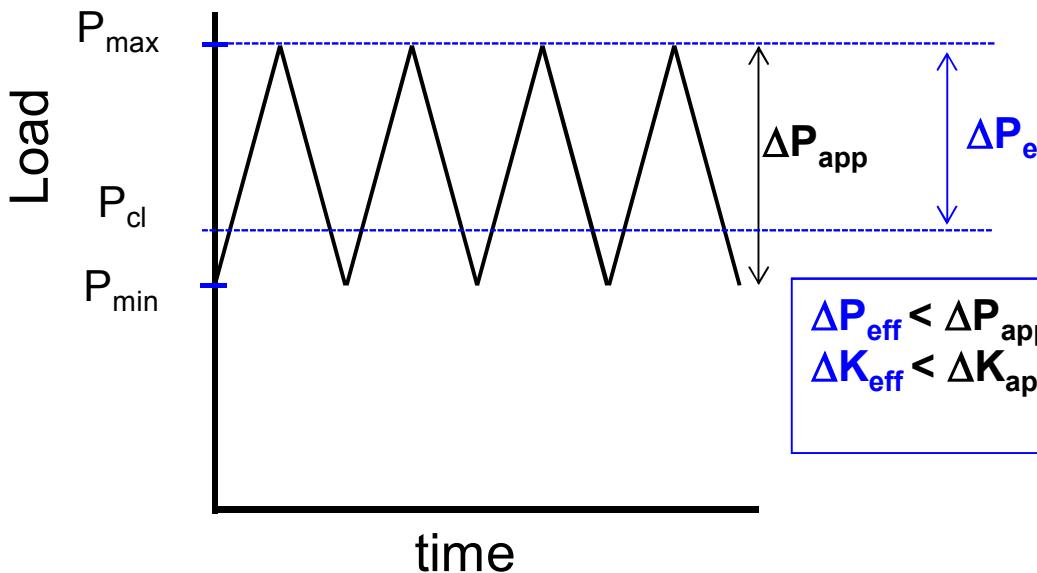
$$\text{Compliance offset} = \frac{[(\text{open} - \text{crack compliance}) - (\text{compliance}) * 100]}{(\text{open} - \text{crack compliance})}$$

Decreased  $\Delta K_{\text{eff}}$  may account for reduced FCGR in HAZ

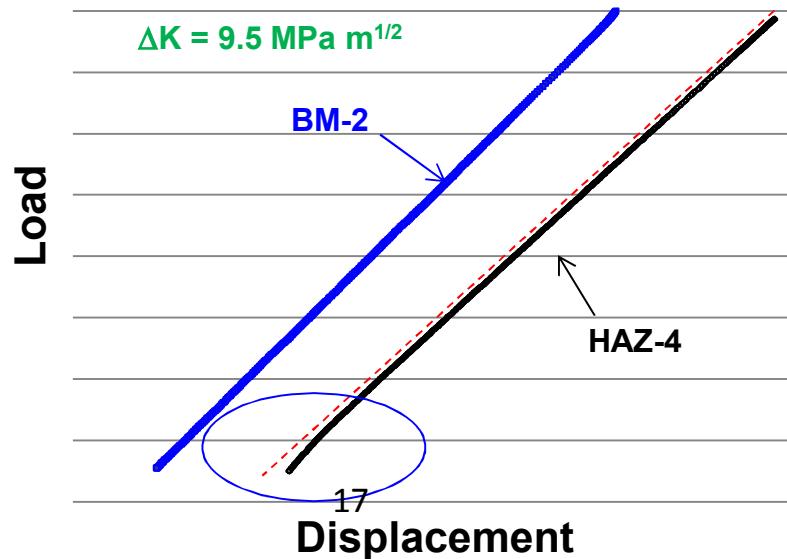
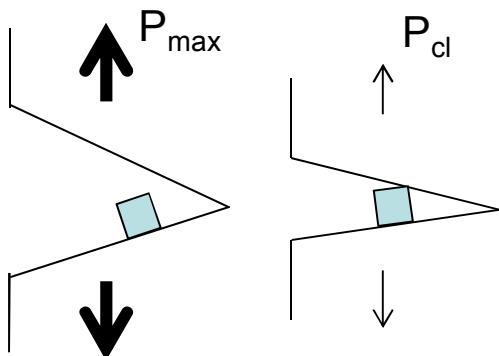
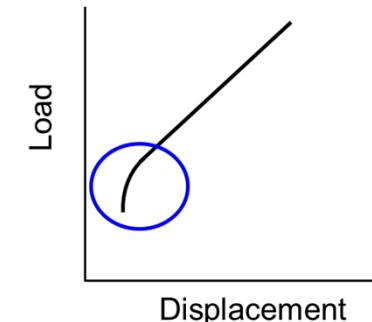
**Possible sources of crack closure**

- Residual stress
- More pronounced Mode II loading

# Crack closure: crack remains in closed position even when tensile load is applied

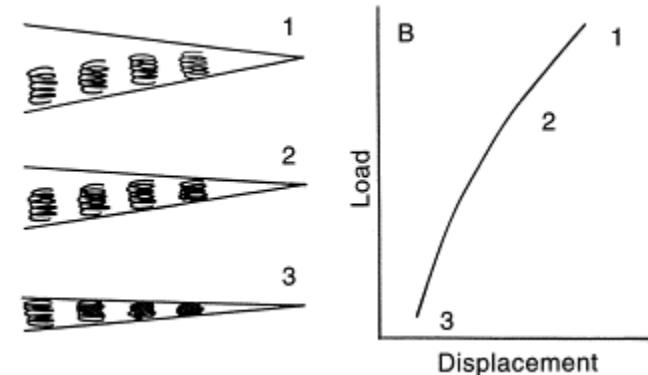
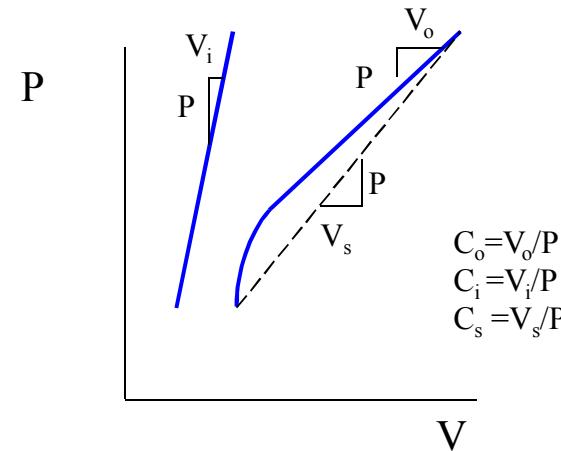
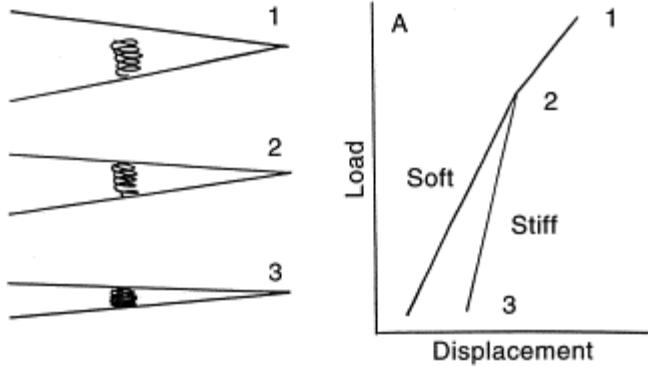


- Below  $P_{cl}$ , contribution to stress intensity is reduced
- Crack closure observed in load-displacement curves as change of slope at lower loads



# Adjusted Compliance Ratio (ACR) was used to correct for crack closure

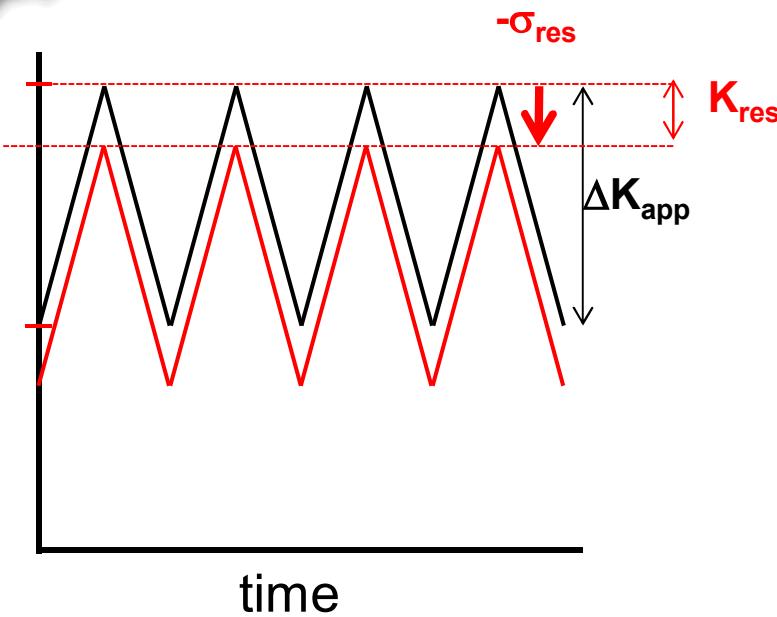
- Alternative method is ASTM 2% compliance offset method
- Both methods are used to determine  $\Delta K_{\text{effective}}$
- $\Delta K_{\text{applied}} > \Delta K_{\text{effective}}^{ACR}$



$$ACR = \frac{C_s - C_i}{C_o - C_i}$$

$$\Delta K_{\text{effective}} = \Delta K_{\text{app}} \times ACR$$

# Compressive residual stress can reduce $K_{max}^{eff}$



$$K_{max}^{eff} = K_{max} + K_{res}$$

- Calculate  $K_{res}$  from load-disp curves

