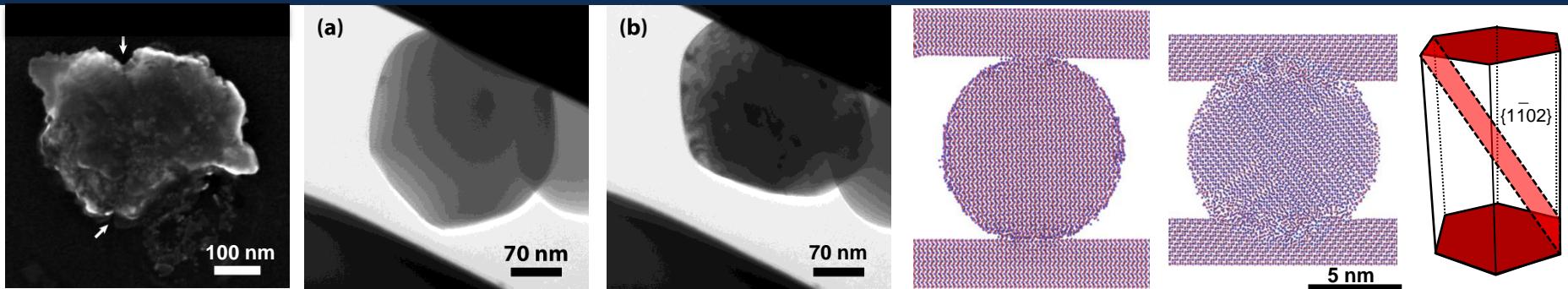


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## Deformation of Alumina Particles in Compression – Basis for a Room Temperature Ceramic Coating Deposition

**Pylin Sarabol, Michael Chandross, Daniel C. Bufford, Khalid Hattar, Paul G. Kotula,  
Brad L. Boyce, Jay D. Carroll, Aaron C. Hall, and William M. Mook**

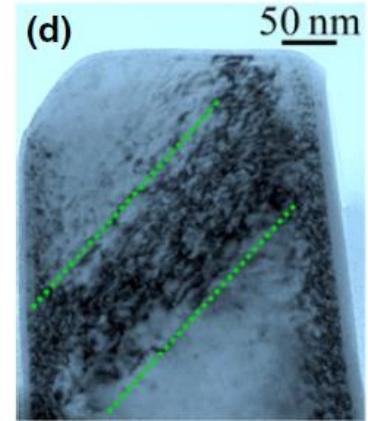
National Lab Day @ Purdue University, Oct 6-7, 2015



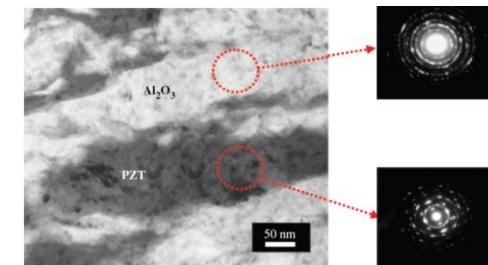
Sandia National Laboratories is a multi-program laboratory managed and operated by Sandia Corporation, a wholly owned subsidiary of Lockheed Martin Corporation, for the U.S. Department of Energy's National Nuclear Security Administration under contract DE-AC04-94AL85000. SAND2015-3704C

# Motivation

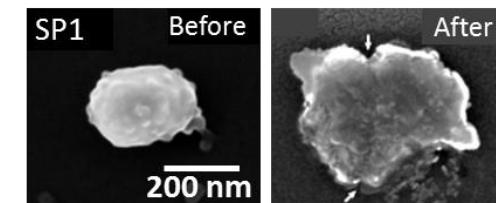
- Bulk materials with high degrees of covalent/ionic bonding, e.g. ceramics, typically undergo brittle fracture when strained.
  - a combination of limited fracture toughness and preexisting flaws.
  - The role of pre-existing flaws and defects can evolve as the characteristic length scale of materials decrease (e.g. micro-pillars and particles) [1-14].
  - In bulk ceramics → crack initiation sites.
- **At small length scales, significant plasticity observed in ceramic single crystals at room temperatures.**
  - Low strain rates → dislocation slip and shape change
    - compressed sapphire micro-pillars [10], particles [16], and confined zones underneath an indenter [36] at RT.
  - **High strain rates → aerosol deposition (AD)**
    - < 2  $\mu\text{m}$  particles are accelerated to high velocity (200-600 m/s) by pressurized gas, impacted, deformed, and consolidated on the substrates under vacuum [16-24].
- Room temperature plasticity in ceramics at small length scale gave insights into future development of alternative ceramic forming technology and high strength/high toughness functional ceramics.
- The focus of this study is to better understand the deformation behavior observed in small-scale, compressed ceramic particles, specifically sapphire or  $\alpha\text{-Al}_2\text{O}_3$  and how they play a role in making AD coatings.



Dislocations on {001} planes in compressed ZrC pillar from S. Kiani, et al. *J. Am. Ceram. Soc.*, 2015;98:2313



AD Al<sub>2</sub>O<sub>3</sub> and PZT composite film from J. Akedo. *J. Am. Ceram. Soc.*, 2006;89:1834



Compressed sapphire particle from P. Sarabol, et al., *JTST*, 2016:25

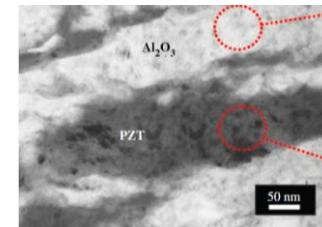
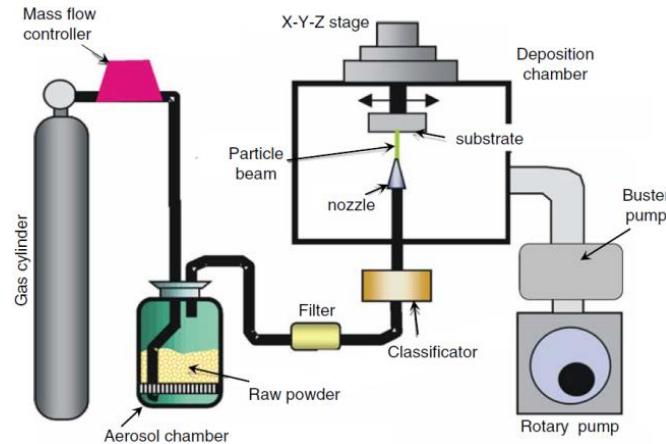
# Motivation

## Aerosol Deposition (AD) enables materials integration.

- Ceramics are conventionally processed at **2000°C**.
- AD process at **room temperature (RT)** in vacuum
  - sub-micron particles accelerated to high velocity by pressurized gas, impacted, consolidated to form a film.
- Similar AD ceramic film microstructures
  - sub-micron particles undergo *plastic deformation*
  - break up into *small crystallites* (20-75 nm)<sup>1-3</sup>
  - planar *defects* and *amorphous regions*<sup>4</sup>.

## Particle deformation/bonding not well understood

- Common deformation mechanisms exist.
- Examine sub-micron ceramic particles RT deformation as a building block for AD coatings.



AD process and coatings from Akedo *J. Am. Ceram. Soc.*, 2006;89:1834



AD Flexible electronics from J. Akedo. *JTTEE5*, 2007:17:181



AD magnetic films from Mizoguchi et al. *J. Magnetic Soc Japan* 2006;30:659

[1] Akedo, J. and Ogiso, H., *JTST*, Vol. 17, (2008), pp. 181-198.  
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[3] Akedo, J. *J. Am. Ceram. Soc.*, Vol. 89, (2006), pp. 1834-1839.  
 [4] Park, H. et al. *Scripta Materialia*, 2015.

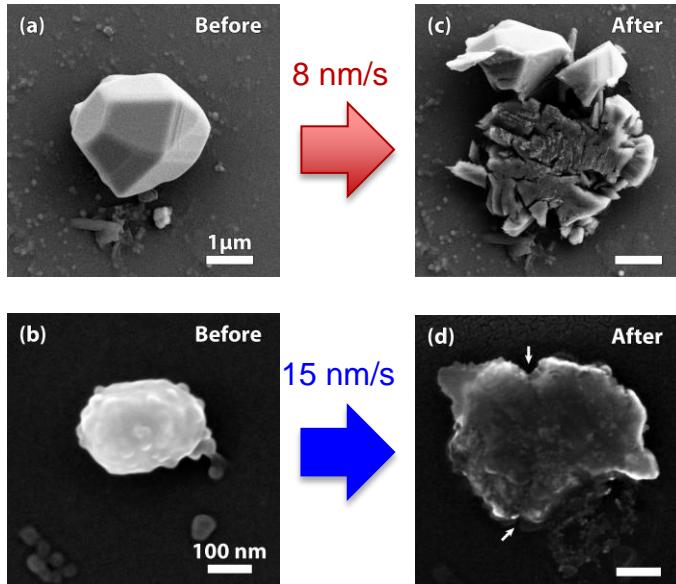
# Previous work – P. Sarobol, et al., JTST., 2016:25

DOI 10.1007/s11666-015-0295-2



- Performed micro-compression on 3.0  $\mu\text{m}$  and 0.3  $\mu\text{m}$   $\text{Al}_2\text{O}_3$  particles
- Micron sized particles - **brittle fracture**
  - Absorbed strain energy density before fracture  **$107 \pm 69 \text{ MJ/m}^3$**
  - Strain before fracture  **$5.5 \pm 1\%$**
- Sub-micron sized particles - **substantial plastic deformation** before fracture.
  - Absorbed strain energy density before fracture  **$630 \pm 238 \text{ MJ/m}^3$**
  - Strain before fracture  **$18 \pm 9\%$**
  - **Deformable sub-micron sized particles = AD coating building block**

- **6x** higher strain energy density
  - dislocation nucleation
- **3x** higher accumulated strain



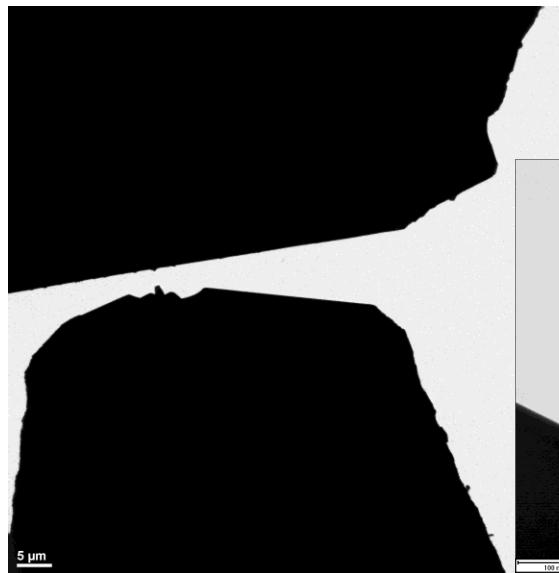
	Micron	Sub-micron
# Pre-existing Defects	High	Moderate
Energy Density Input	Low	Moderate
Governing Mechanism(s)	Fracture	Plasticity + Fracture
Response to Compression	Crack initiation & Propagation	Dislocation nucleation, slip, crack initiation & propagation

***Sub-micron sized  $\text{Al}_2\text{O}_3$  response to compression***  
– *In Situ TEM* micro-compression  
– Molecular Dynamics Simulation

# In Situ TEM Compression

## In Situ Micro-Compression<sup>5</sup> – 300 nm particles

- Single crystal, ultra pure 300 nm sapphire ( $\alpha$ -Al<sub>2</sub>O<sub>3</sub>) particles.
- A Hysitron PI95 TEM Picoindenter with a 1  $\mu$ m diameter flat punch tip and the a JEOL 2100 LaB<sub>6</sub> TEM<sup>7</sup> at 200 kV were used.
- Compression done in ***open loop*** mode with the loading rate of 10  $\mu$ N/s (approx. < 2 nm/s displ rate). Images taken at 15 fps.

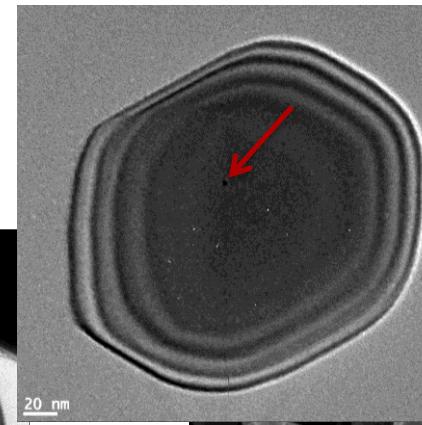


*In situ* TEM micro-compression on 0.3 $\mu$ m particle

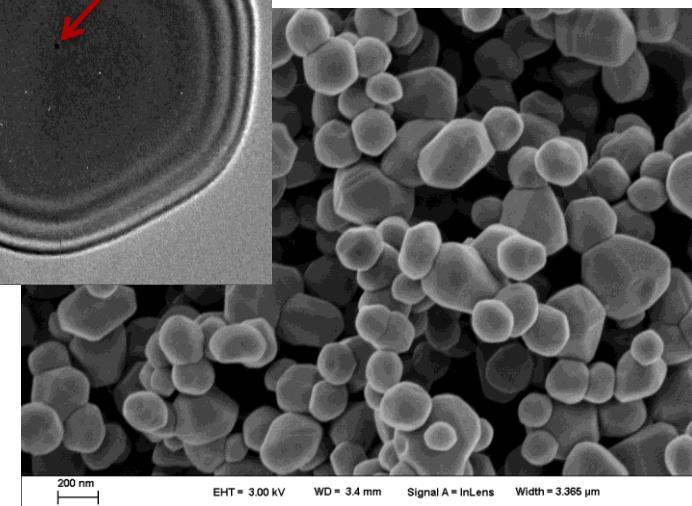
[5] Sarabol, P., et al., SAND2014-18127, (2014).

[6] Hysitron I (2013) SEM Picoindenter User Manual. Revision 9.3.0913 edn.

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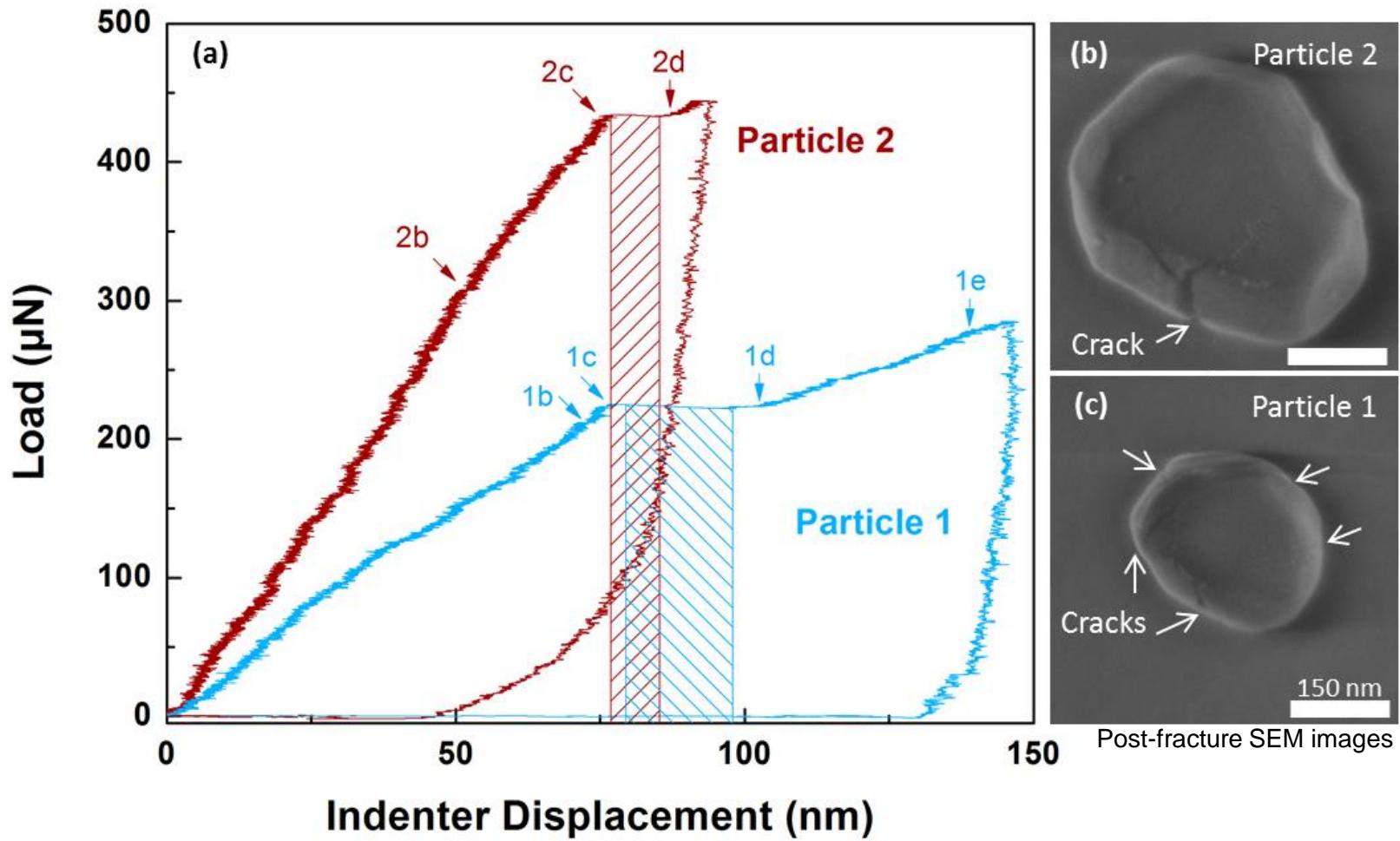


Bright field TEM image of a 300 nm particle oriented on the [001] zone axis.



SE SEM image of the 300 nm

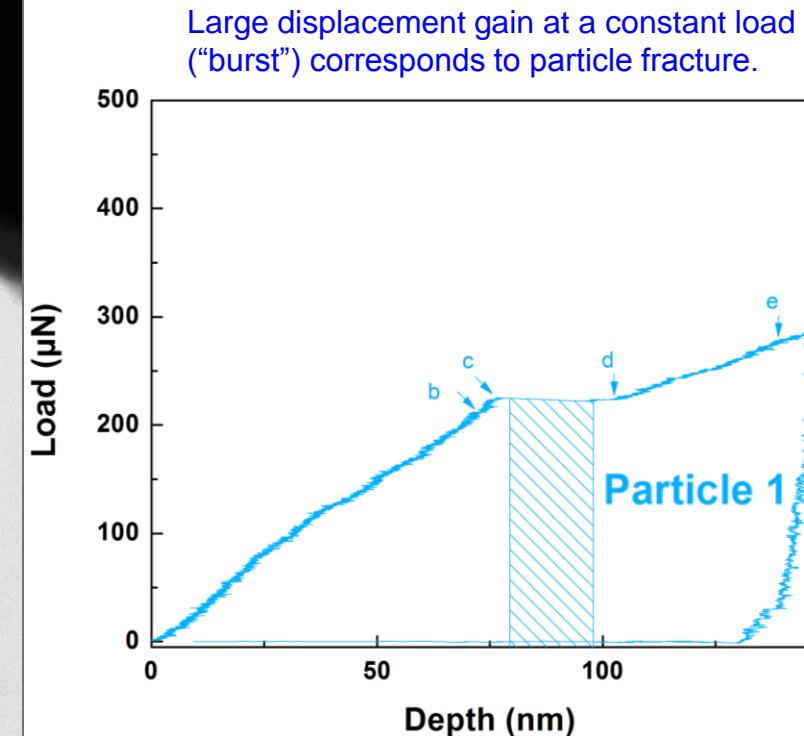
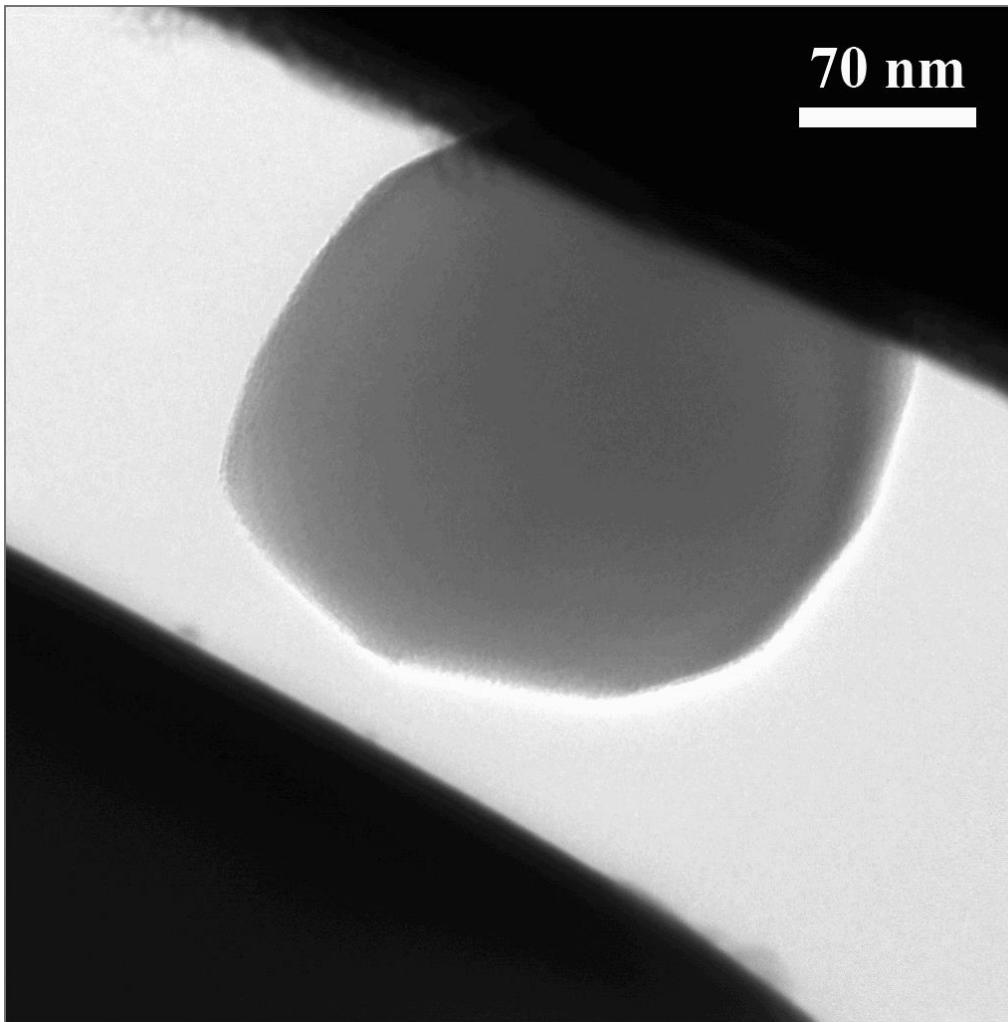
# In Situ TEM Compression



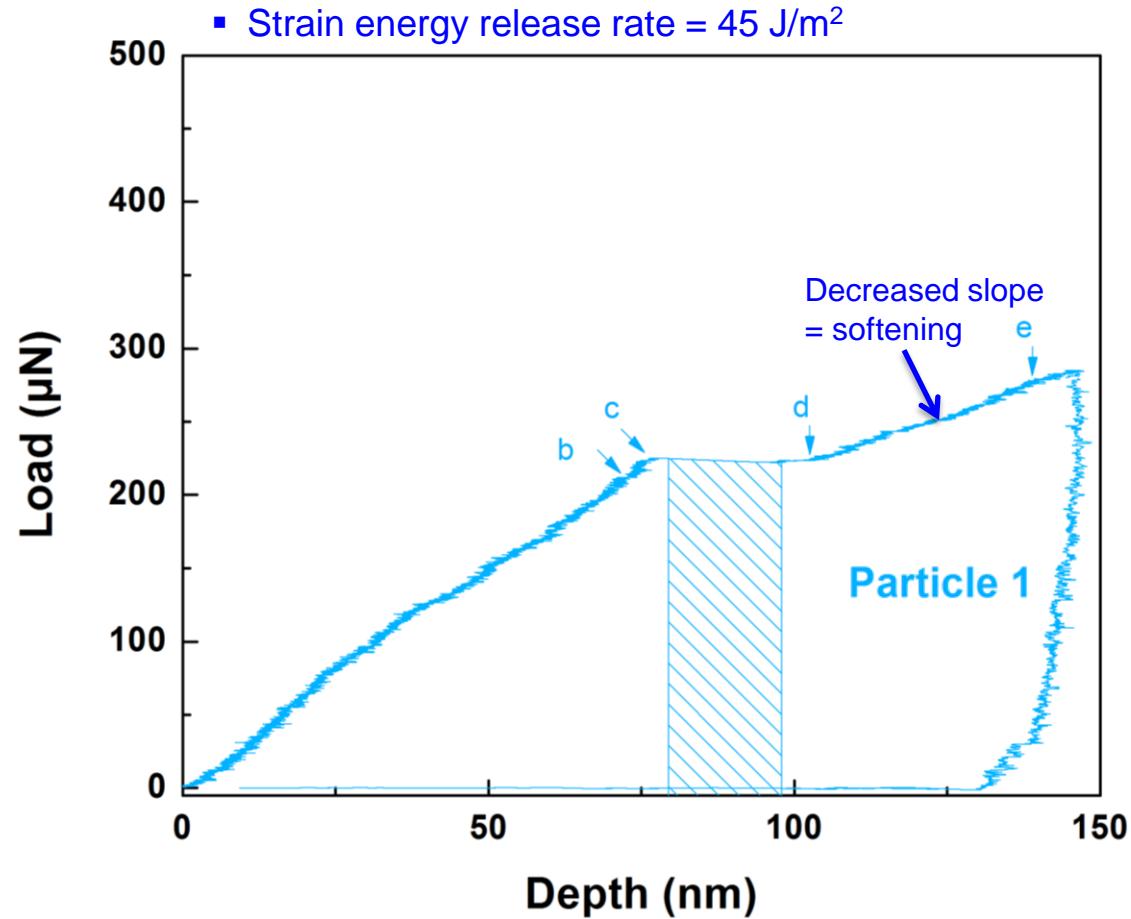
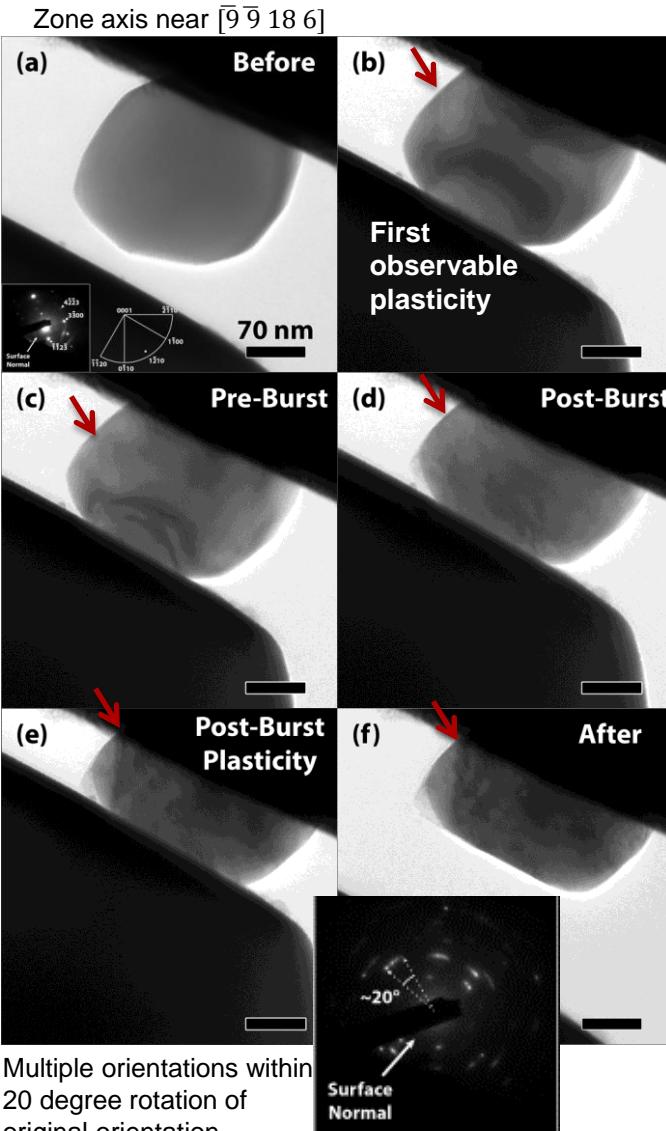
- Elastic to Plastic transitions are unclear. Seemed to happen much earlier in the loading (first 5-10 nm displacement). Absence of concavity and linearity of the curves were surprising.
- $G_C$  values for Particle 1 and 2 are 45 J/m<sup>2</sup> and 17 J/m<sup>2</sup>, respectively. Values within the calculated range of orientation-dependent  $G_C$  of single crystal alumina of 16 - 65 J/m<sup>2</sup> [47].

# *In Situ* TEM Compression – P1

Diameter  $\sim 0.24 \mu\text{m}$ , Open loop, Strain rate  $\sim 0.009 \text{ s}^{-1}$



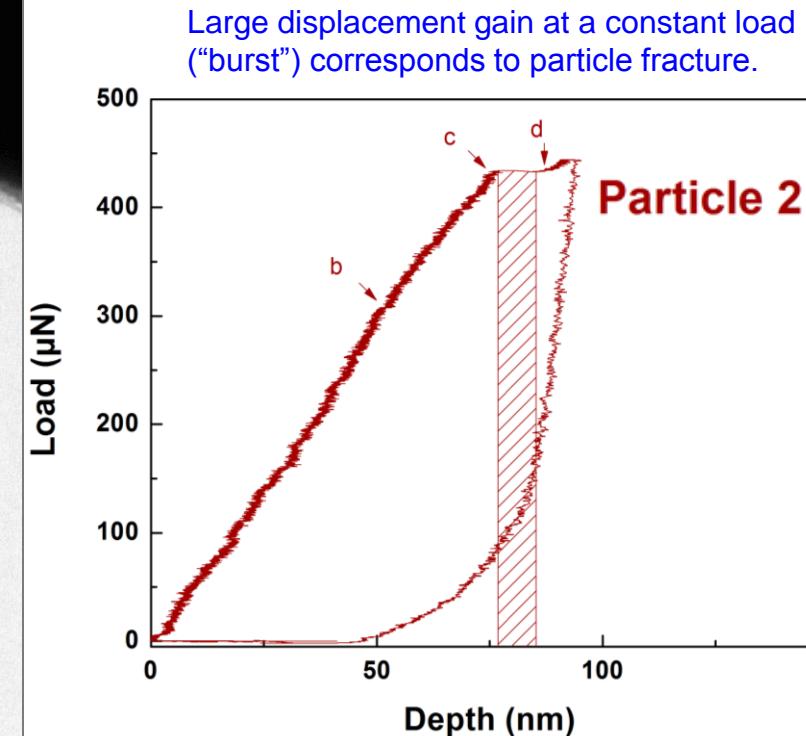
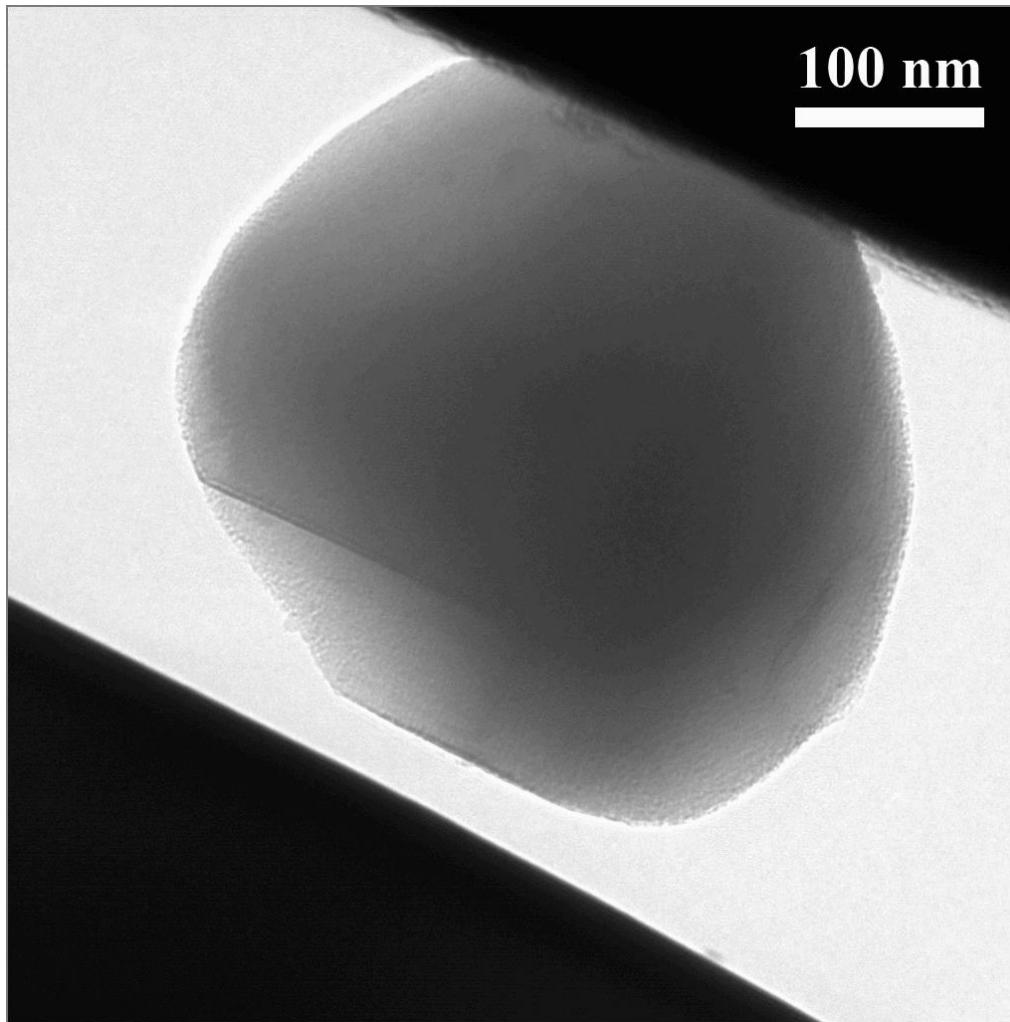
# In Situ TEM Compression – P1



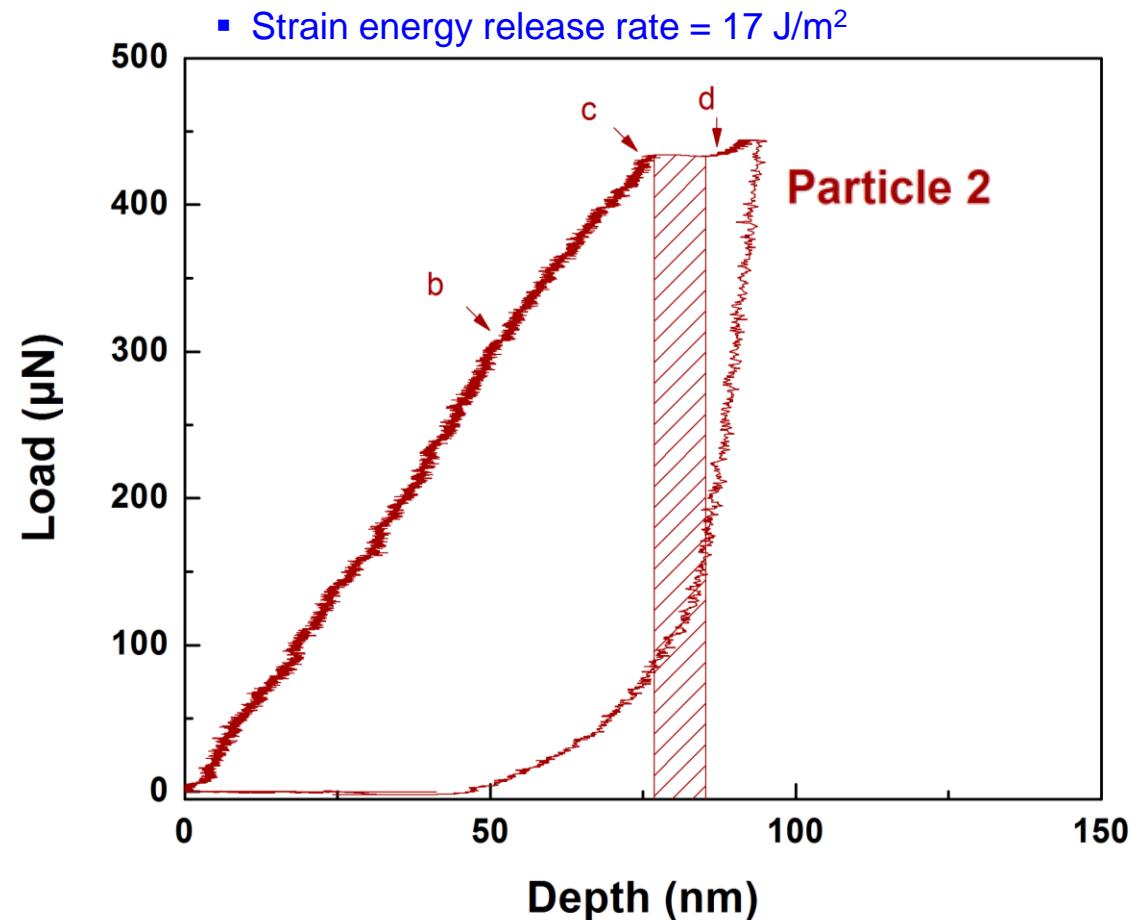
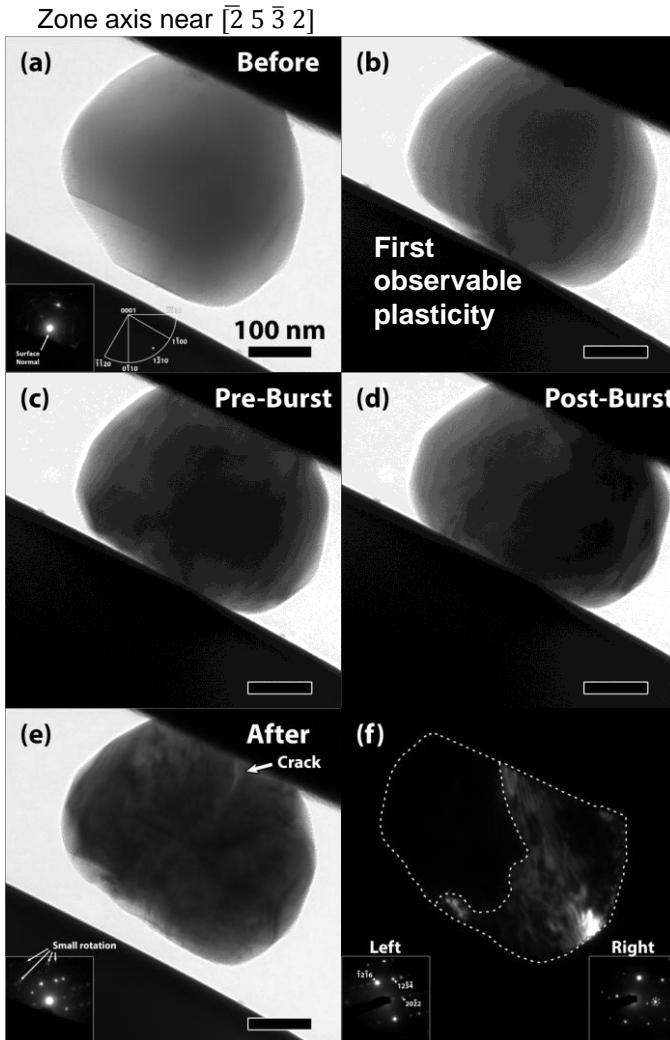
- Pre-burst plasticity: small regime with low dislocation activity.
- **Crack nucleation and propagation** leading to through-particle fracture.
- Post-burst plasticity: high dislocation activities, change in deformation mechanism as indicated by lower slope.
- **Mosaicity** with a 20 degree orientation spread.

# *In Situ* TEM Compression – P2

Diameter  $\sim 0.38 \mu\text{m}$ , Open loop, Strain rate  $\sim 0.005 \text{ s}^{-1}$



# In Situ TEM Compression – P2



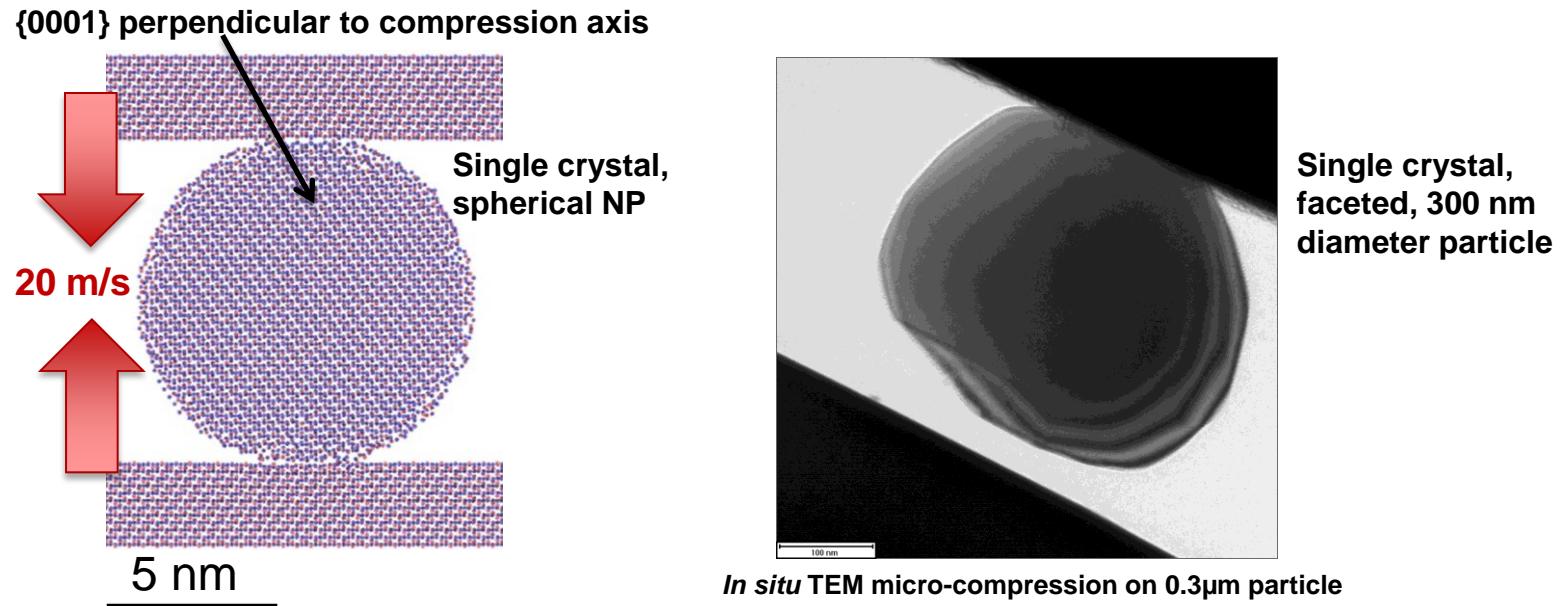
Two halves related by slight rotation, both near  $\bar{1} 2 1 6$  zone axis

- Pre-burst plasticity: large regime with **high dislocation activity** (nucleation and moving through particle).
- **Crack nucleation and propagation** leading to through-particle fracture.

# Simulated Particle Compression

## Molecular Dynamics Simulations – 10 nm nanoparticles (NPs)

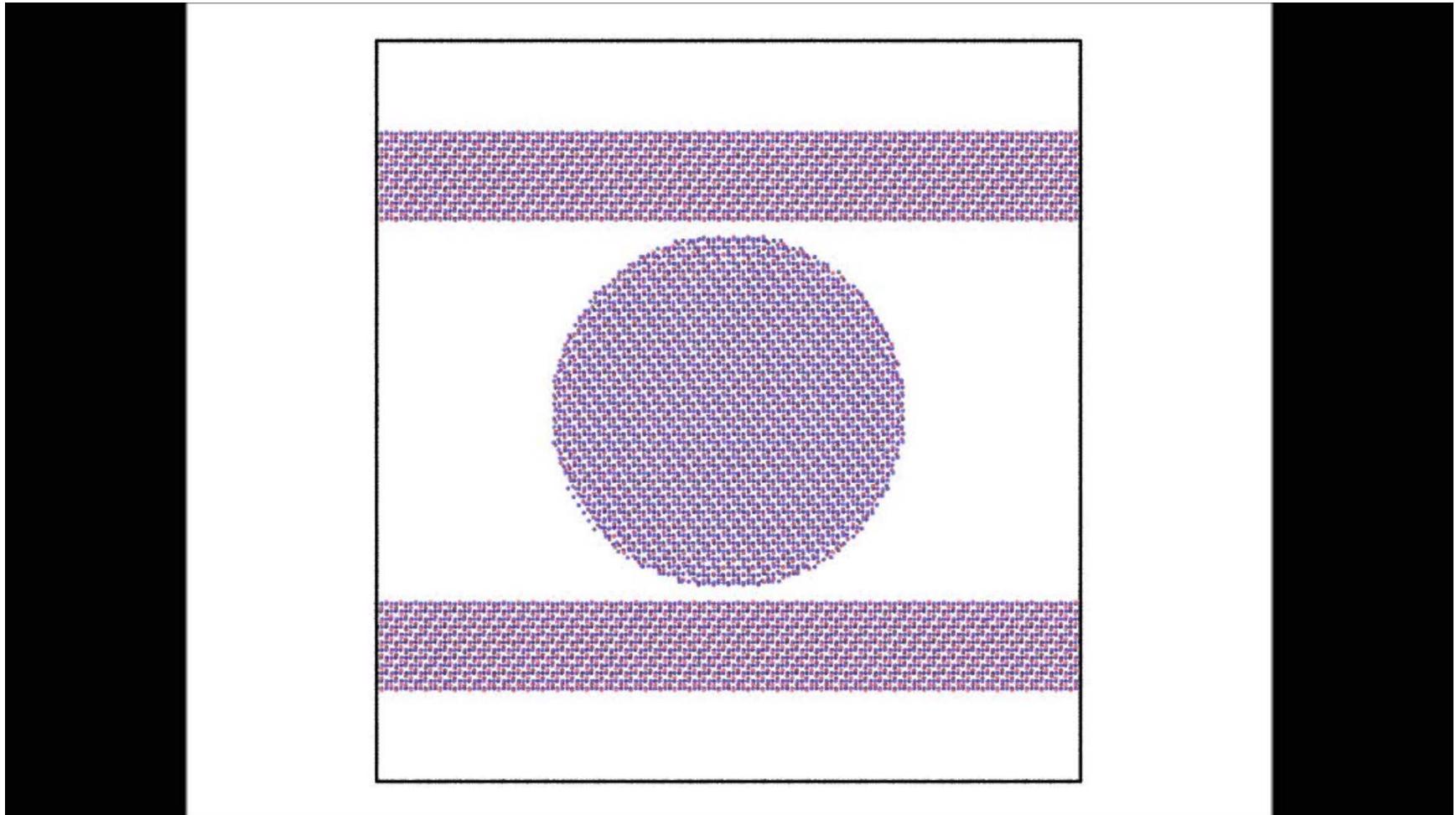
- MD allows identification of dislocations, slip planes, and particle fracture.
- Long computing time to simulate size > 50 nm (~36 million atoms)
- **Simulating 10 nm sapphire nanoparticle (NP) (~300,000 atoms)**
- A force-field for ceramics, developed by Garofalini<sup>8</sup>.
- NPs were compressed (by ~1/3 of the initial diameter) between sapphire (single crystal  $\alpha\text{-Al}_2\text{O}_3$ ) walls at a constant velocity of **20 m/s**. “**Displacement control**”.



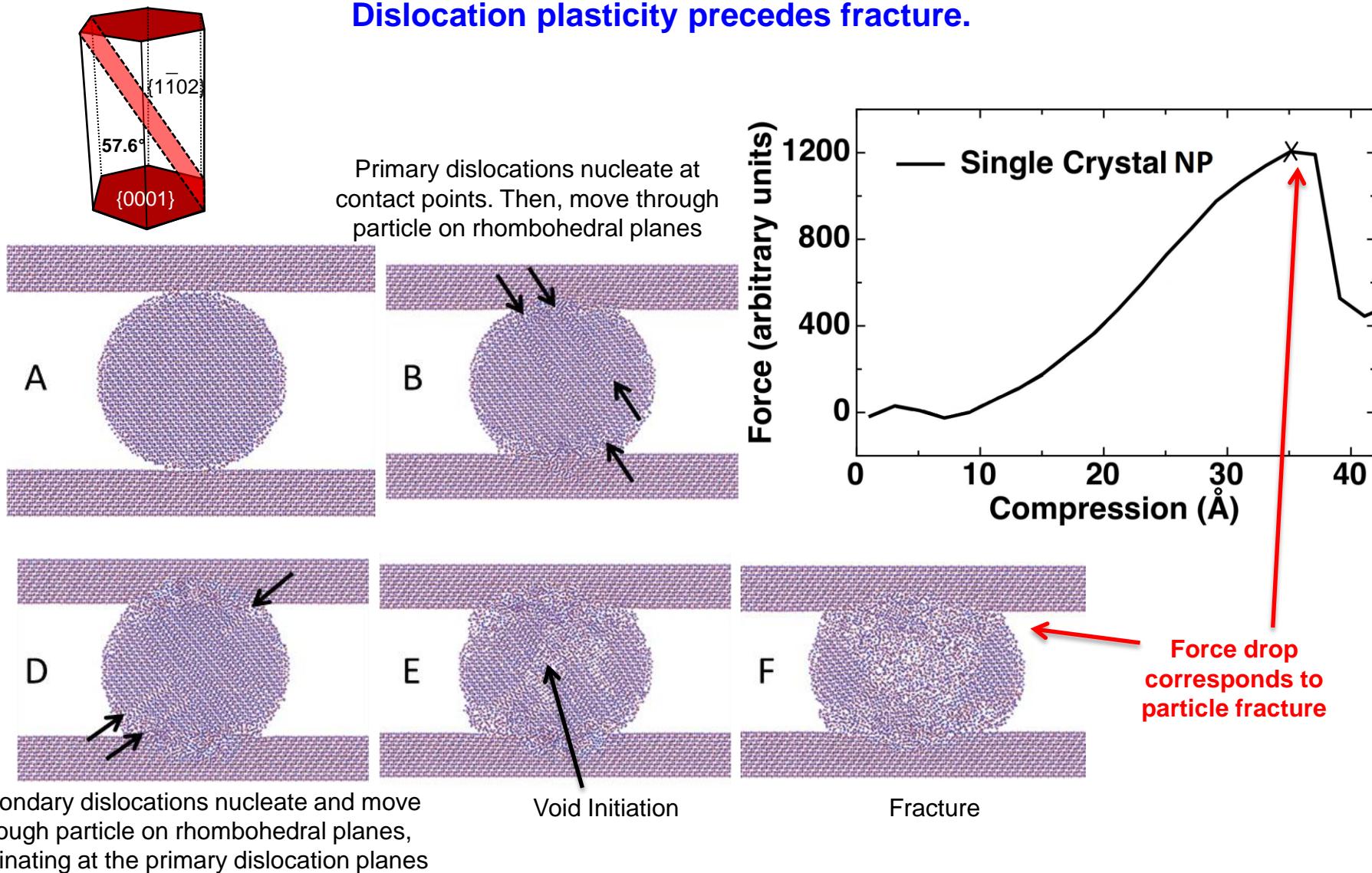
# MD Simulation Results

10 nm diameter, defect-free, single crystal  $\alpha$ -alumina, compression axis  $\perp (0001)$

20 m/s  $\rightarrow$  dislocation nucleation and glide on Rhombohedral planes then fracture



# MD Simulation Results



# Conclusions

- The findings from *in situ* TEM micro-compression experiments and molecular dynamic simulations agree well:
  - Dislocation plasticity precedes fracture in compressed small sapphire particles at RT.
  - Range of responses to compression includes
    - Dislocation nucleation, slip, movement
    - Significant shape change
    - Orientation spread (mosaicity)
    - Fracture
- Use info to inform feedstock preparation, aerosol deposition parameters, and particle-particle bonding in the consolidated coatings.
- Room temperature plasticity in ceramics at small length scale gave insights into future development of alternative ceramic forming technology and high strength/high toughness functional ceramics.

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Thank you,  
for your attention.  
Pylin Sarabol –  
psarobo@sandia.gov

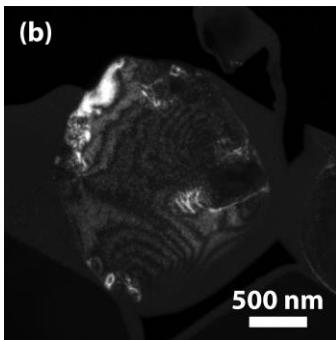
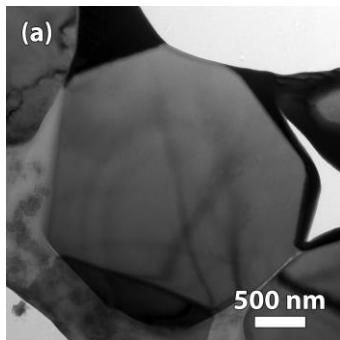
# BACK UP SLIDES

# Ceramic Particle RT Deformation - Sapphire

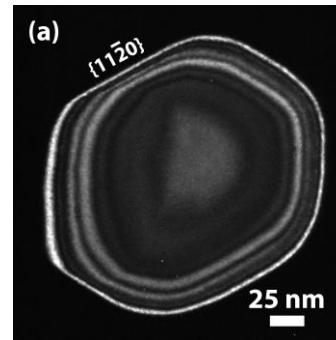
- Deformation behavior influenced by *number of internal defects*, temperature, crystal orientation/size. Numbers of pre-existing (immobile) defect scale with size.
- In situ SEM/TEM micro-compression and Molecular Dynamics Simulations

Proposed

	Micron	Sub-micron
# Pre-existing Defects	High	Moderate
Energy Density Input	Low	Moderate
Governing Mechanism(s)	Fracture	Plasticity + Fracture
Response to Compression	Crack initiation & Propagation	Dislocation nucleation, slip, crack initiation & propagation
Compression Testing	SEM	SEM and TEM



3.0 $\mu\text{m}$  Highly Defective



0.3 $\mu\text{m}$  Nearly Defect Free

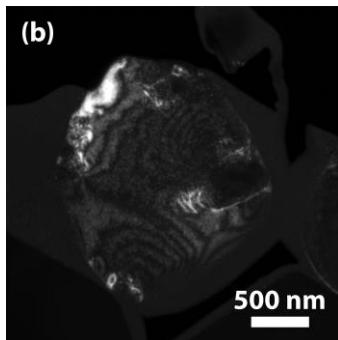
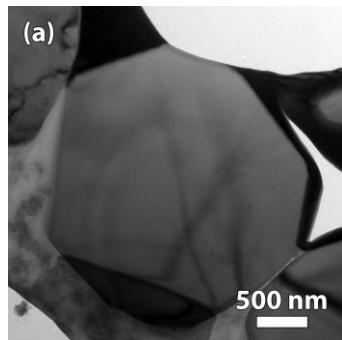
- Infeasible (long computing time) to perform molecular dynamics simulations on size  $>0.05\mu\text{m}$
- 'smaller' particles ( $0.3\mu\text{m}$ ) are nearly defect-free, and 'larger' particles ( $3.0\mu\text{m}$ ) contain immobile defects that serve as crack nucleation sites.
- Circumvented the size limitation of our models by simulating similar sized (10 nm) nanoparticles (NPs) that were either
  - single crystal
  - contained a grain boundary (GB) as an initial immobile defect.
- This approach still enables the study of NP deformation/fracture in computationally-feasible systems.

# Ceramic Particle RT Deformation - Alumina

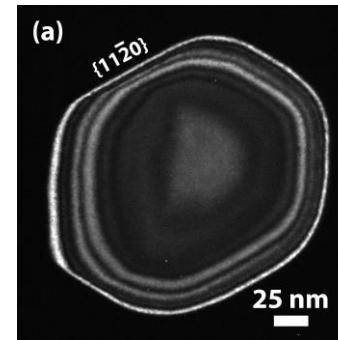
- Deformation behavior influenced by *number of internal defects*, temperature, crystal orientation/size. Numbers of pre-existing (immobile) defect scale with size.
- In situ SEM/TEM micro-compression and Molecular Dynamics Simulations

Proposed

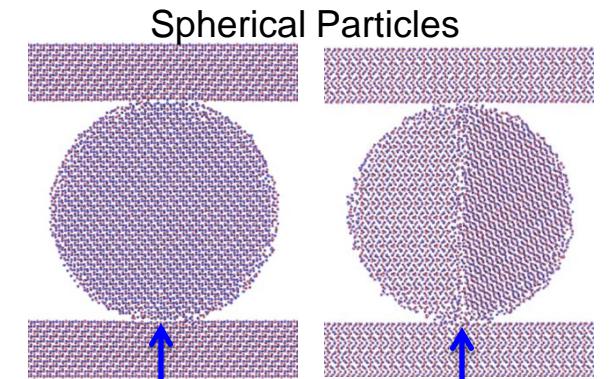
	Micron	Sub-micron	Single Crystal Nano	Bicrystal Nano
# Pre-existing Defects	High	Moderate	None	Grain Boundary
Energy Density Input	Low	Moderate	High	Low
Governing Mechanism(s)	Fracture	Plasticity + Fracture	Plasticity	Fracture
Response to Compression	Crack initiation & Propagation	Dislocation nucleation, slip, crack initiation & propagation	Dislocation nucleation, Slip	Crack initiation & propagation
Compression Testing	SEM	SEM and TEM	MD Simulation	MD Simulation



3.0 $\mu$ m Highly Defective



0.3 $\mu$ m Nearly Defect Free

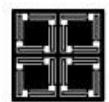
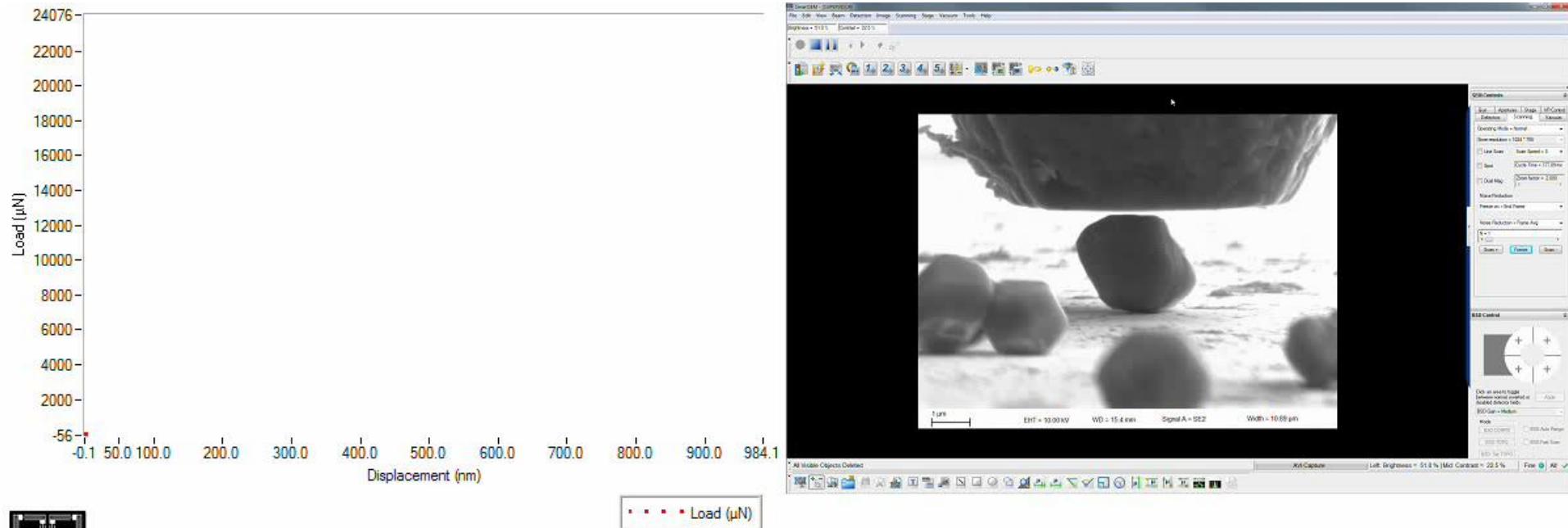


10 nm Defect Free 10 nm with a GB 18

# *In Situ* SEM micro-compression – 3.0 $\mu\text{m}$



Displacement control, Strain rate  $\sim 0.003 \text{ s}^{-1}$



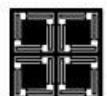
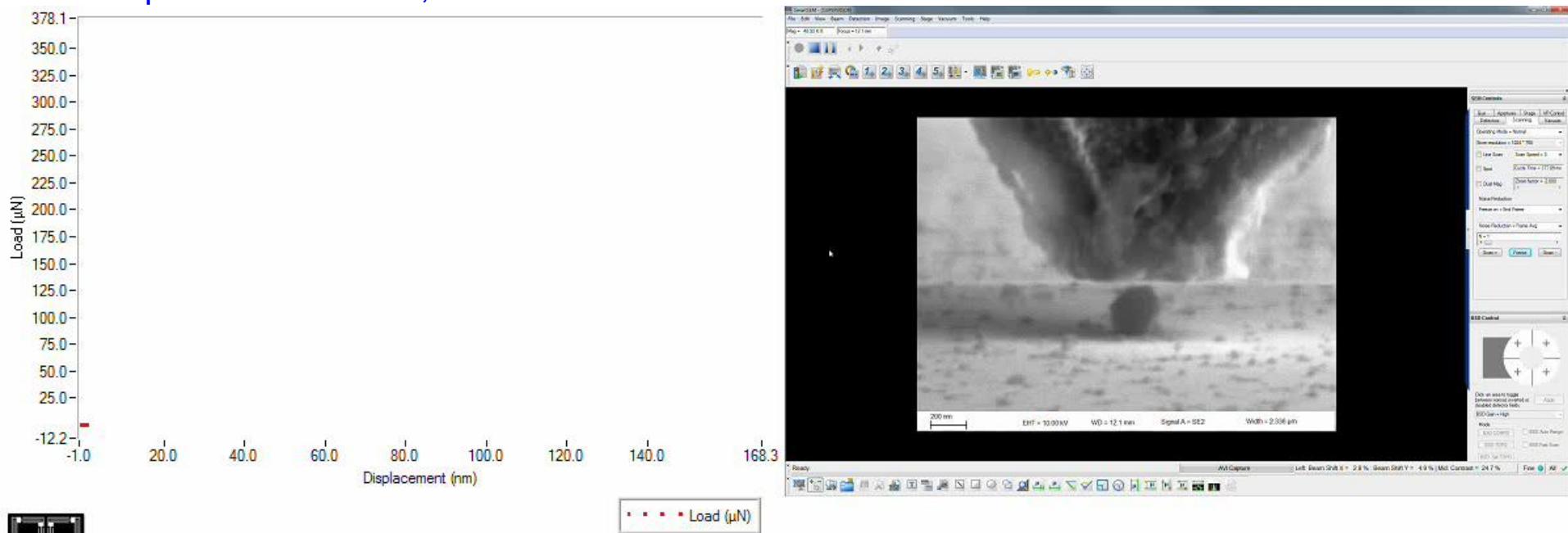
HYSITRON™

- Compressed 4 particles
- No observable shape change prior to fracture and fragmentation
- Displacement excursion corresponded to a fast fracture event
  - Strain Energy Density before Fracture  $\sim 203 \text{ MJ/m}^3$
  - Strain at fracture  $\sim 7\%$

Tip could not keep up with large displacement gained during fracture.

# In Situ SEM micro-compression – 0.3 $\mu$ m

Displacement control, Strain rate  $\sim 0.05 \text{ s}^{-1}$

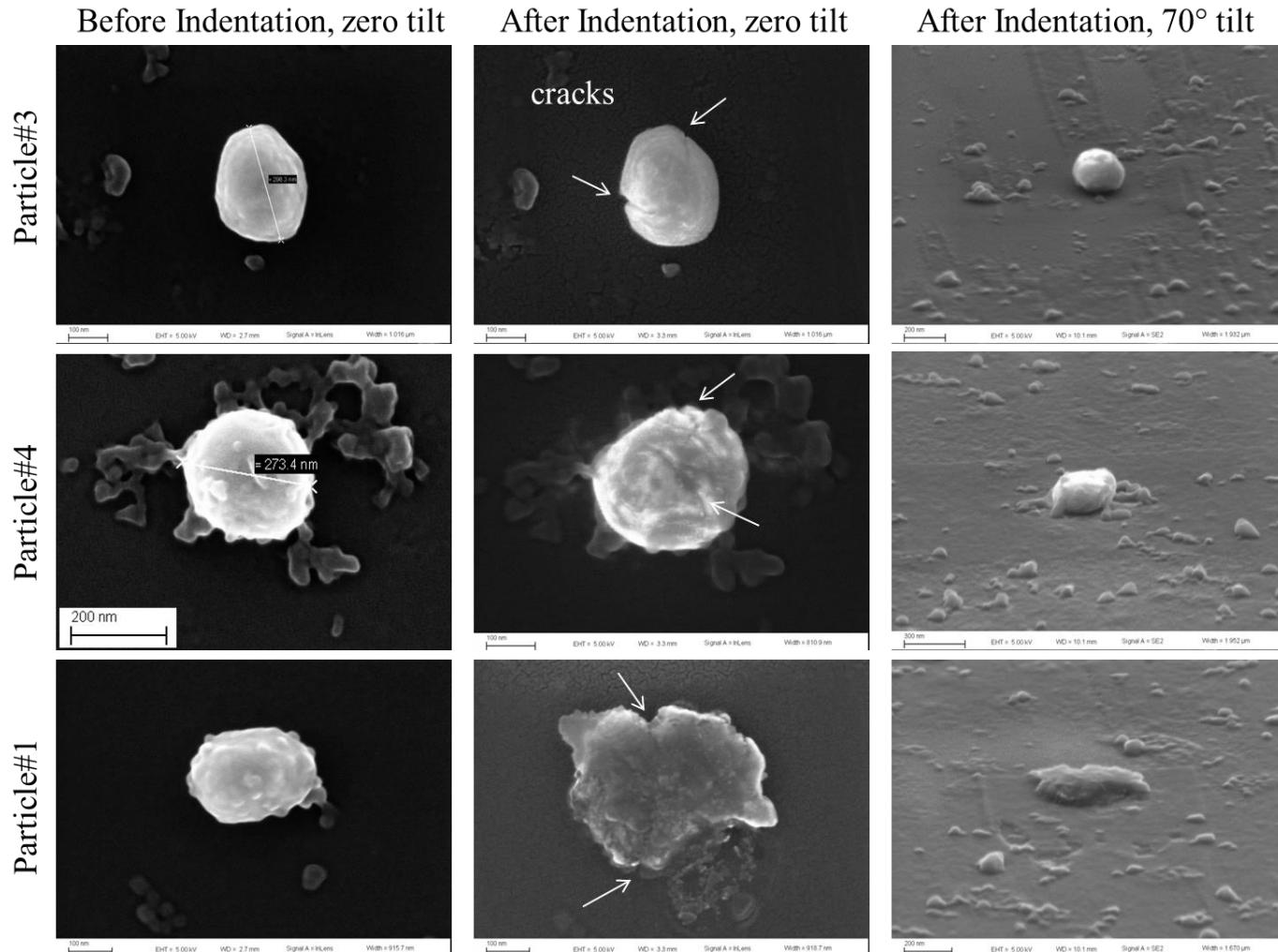


HYSITRON™

- Compressed 4 particles
- Significant plastic deformation/ shape change and stayed intact
- Displacement excursion corresponded to??? *Ex situ* observation
- Strain Energy Density before displacement excursion  $\sim 675 \text{ MJ/m}^3$ 
  - Strain at displacement excursion  $\sim 16\%$

Tip could not keep up with large displacement gained during fracture.

# Ex Situ SEM observation – 0.3 $\mu$ m



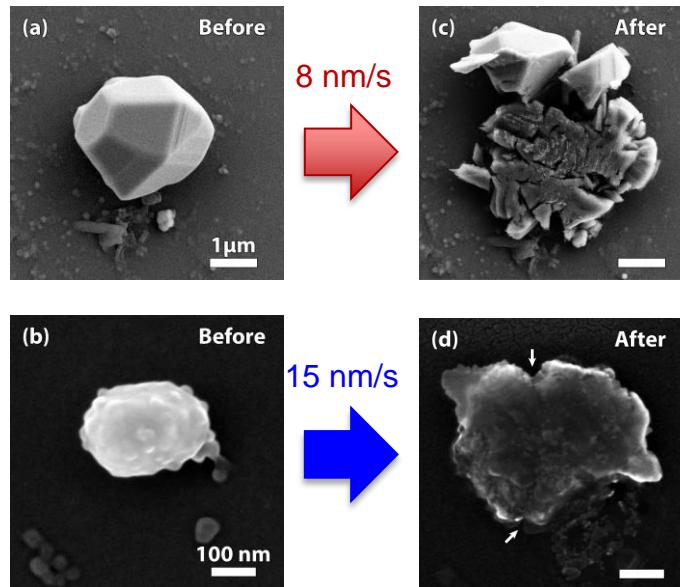
~307  $\mu$ N  
Max load

~420  $\mu$ N  
Max load

Extreme  
Loading

Different deformation behavior and load at first fracture may differ from particle-to-particle due to orientation differences and different pre-existing defect densities. However, overall, the sub-micron sized alumina particles exhibited significant plastic deformation before fracture.

# Micro-compression Summary



Particle Identifier	Diameter (μm)	Nominal Strain Rate (s <sup>-1</sup> )	Strain Energy Density Before Displacement Excursion (MJ/m <sup>3</sup> )	Strain at displacement excursion (%)
<b>Large Particles</b>				
SEM-LP1	2.9	0.03	47	5
SEM-LP2	2.6	0.006	106	5
SEM-LP4	2.9	0.005	70	5
SEM-LP5	2.9	0.003	203	7
<b>Avg Large Particles</b>	<b>2.8</b>	-	<b>106±69</b>	<b>5.5 ± 1</b>
<b>Small Particles</b>				
SEM-SP2	0.17	0.09	494	11
SEM-SP3	0.29	0.05	366	12
SEM-SP4	0.28	0.05	607	13
SEM-SP5	0.29	0.05	675	16
*TEM-SA2	0.38	*0.005	573	32
*TEM-SB1	0.24	*0.009	1066	27
<b>Avg Small Particles</b>	<b>0.26</b>	-	<b>630±238</b>	<b>18 ± 9</b>

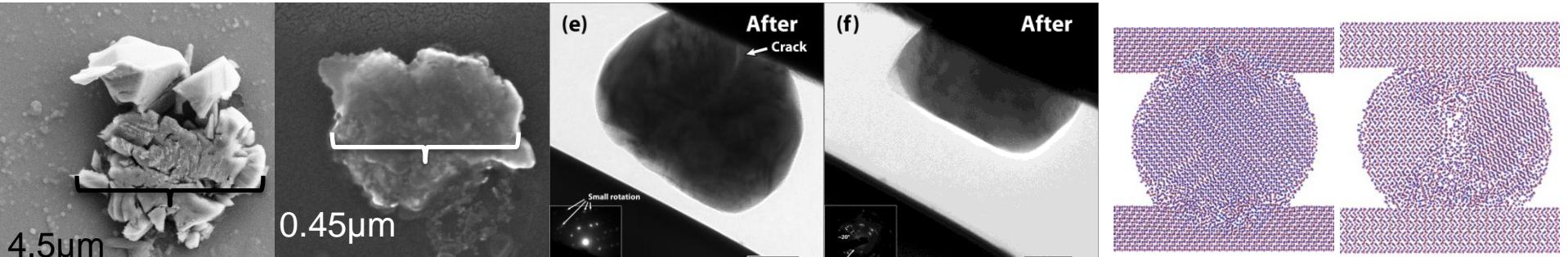
	Micron	Sub-micron
# Pre-existing Defects	High	Moderate
Energy Density Input	Low	Moderate
Governing Mechanism(s)	Fracture	Plasticity + Fracture
Response to Compression	Crack initiation & Propagation	Dislocation nucleation, slip, crack initiation & propagation

- Micron sized particles - brittle fracture
- Sub-micron sized particles - substantial plastic deformation before fracture and/or coordinated shear deformation.
  - **6x** higher strain energy density input
  - dislocation nucleation
- **3x** higher accumulated strain
- In some cases, became polycrystalline.

# Ceramic Particle RT Deformation - Alumina

- Deformation behavior influenced by **numbers of internal defects, orientation, size.**

	Micron	Sub-micron	Single Crystal Nano	Bicrystal Nano
# Pre-existing Defects	High	Moderate	None	Grain Boundary
Energy Density Input	Low	Moderate	High	Low
Governing Mechanism(s)	Fracture	Plasticity + Fracture	Plasticity	Fracture
Response to Compression	Crack initiation & Propagation	Dislocation nucleation, slip, crack initiation & propagation	Dislocation nucleation, Slip	Crack initiation & propagation
Compression Testing	SEM	SEM and TEM	MD Simulation	MD Simulation



**3.0μm - Fracture and Fragmentation**

**0.3μm – plastic deformation, shape change, cracking**

**0.3μm - Dislocation Plasticity & through particle fracture**

**0.3μm - Coordinated Shear Deformation - Polycrystalline**

**10 nm - Coordinated 10 nm - Fracture Shear Deformation**

# MD Simulation Results

10 nm diameter, contain a GB, 'Janus'  $\alpha$ -alumina,  
20 m/s, left side randomly oriented and right side compression axis  $\perp$  (0001)  $\rightarrow$  Fracture

