



# Hydrogen and Helium Isotopes in Materials Conference

SAND2007-0682C

## February 6-7, 2007

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### “Neutron Diffraction of ErD<sub>2</sub> powders”

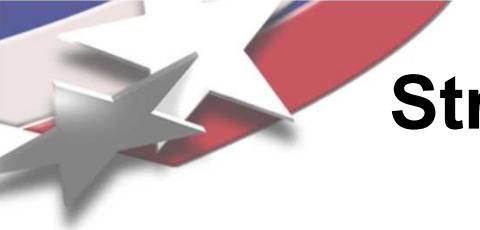
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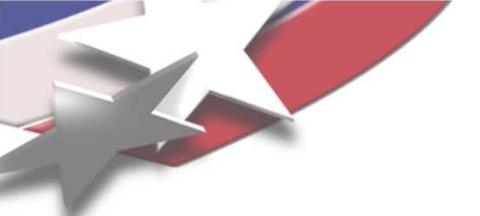


# Structural Analysis of T-site occupancy in $\text{ErT}_2$ films is very challenging

- Difficult determination of (H,D,T) in host lattice via XRD (i.e. only  $1\text{e}^-$ ).
- (D,T) scatter well for neutron diffraction experiments.
- Strong texture of  $\text{ErT}_2$  films complicates structural analysis.
- Loose powders are more random, providing straightforward analysis.
- Loose powder of  $\text{ErT}_2$  is major contamination issue.

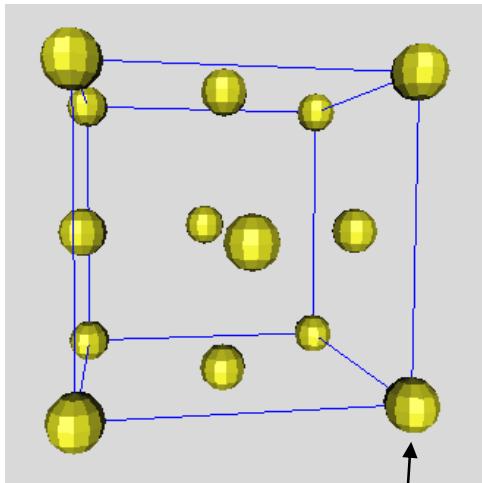
  

- Loose powders of  $\text{ErD}_2$  chosen as surrogate powders for structural analysis of the fluorite lattice.
  - Gain insight into site occupancy without hazards of tritium

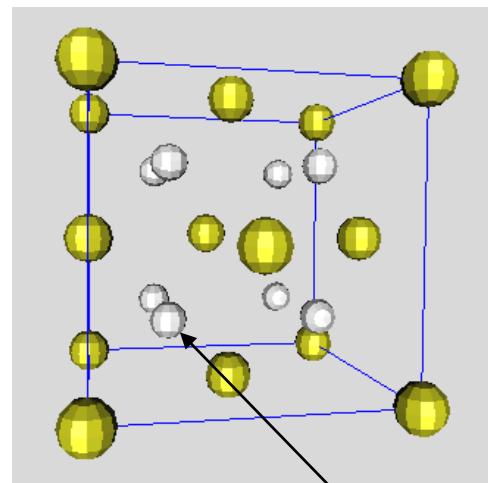


# Fluorite structure of ErD<sub>2</sub>

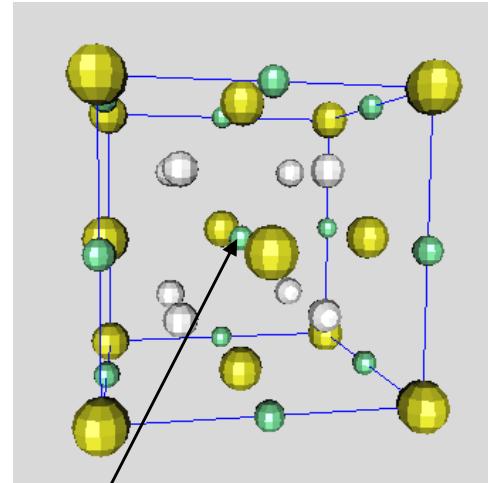
Where is the D going?



FCC lattice of Er



D occupies tetrahedral  
( $\frac{1}{4}, \frac{1}{4}, \frac{1}{4}$ ) sites



Octahedral ( $\frac{1}{2}, \frac{1}{2}, \frac{1}{2}$ )  
sites could also contain D\*

\* T. J. Udovic, J.J. Rush, and I. S. Anderson, *Phys. Rev. B*, **50**, pp.7144 (1994)

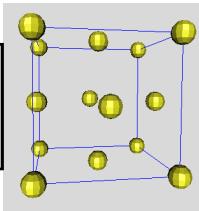


# ErD<sub>2</sub> powder analysis

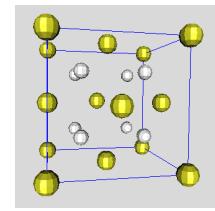
- Er metal D-loaded at various temperatures/pressures.
- ErD<sub>2</sub> powder specimens mixed with silicon powder standard.
- Powders placed in Vanadium canisters.
- Neutron diffraction data collected using HIPD at LANSCE/LANL.
- Structure refinements performed using GSAS software.
- Two run cycles (FY04 and FY05).

# How D changes relative intensities ( $I_{\text{rel}}$ ) for the $\text{ErD}_2$ fluorite-type structure

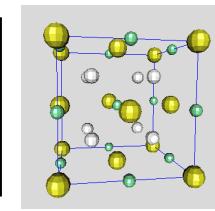
Model 1  
Er only



Model 2  
Er +  $D_{\text{tet}}$



Model 3  
Er +  $D_{\text{tet}}$   
+ 10%  $D_{\text{oct}}$



XRD calculated patterns

$hkl$	d (Å)	$I_{\text{rel}} (\%)$ Model 1	$I_{\text{rel}} (\%)$ Model 2	$I_{\text{rel}} (\%)$ Model 3	Diff % 3-1
111	2.954	100	100	100	
200	2.558	51	49	49	-2
220	1.808	35	36	36	+1
311	1.543	40	40	40	
222	1.477	11	11	11	
400	1.279	5	5	5	
331	1.174	15	15	15	
420	1.144	13	13	13	
422	1.044	10	10	10	

Neutron calculated patterns

$hkl$	d (Å)	$I_{\text{rel}} (\%)$ Model 1	$I_{\text{rel}} (\%)$ Model 2	$I_{\text{rel}} (\%)$ Model 3	Diff. % 3-1
111	2.954	100	26	21	-79
200	2.558	56	6	5	-51
220	1.808	55	100	100	+45
311	1.543	78	20	16	-62
222	1.477	24	3	2	-22
400	1.279	13	24	24	+11
331	1.174	45	12	9	-36
420	1.144	43	5	4	-39
422	1.044	36	67	67	+31

# Neutron Powder Diffraction results - FY04

## ErD<sub>2</sub> powder (450 °C) loading

### Crystal structure data

<u>atom</u>	<u>x,y,z</u>	<u>occ.</u>	<u>B(Å<sup>2</sup>)</u>
Er	0,0,0	1.0	0.15(2)
D <sub>tet</sub>	1/4,1/4,1/4	1.00(1)	1.01(2)
D <sub>oct</sub>	1/2,1/2,1/2	0.02(1)	1.01

$$a = 5.1187(2) \text{ Å}$$

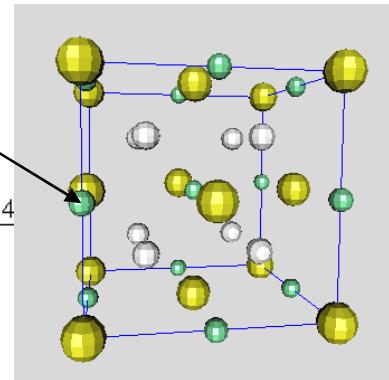
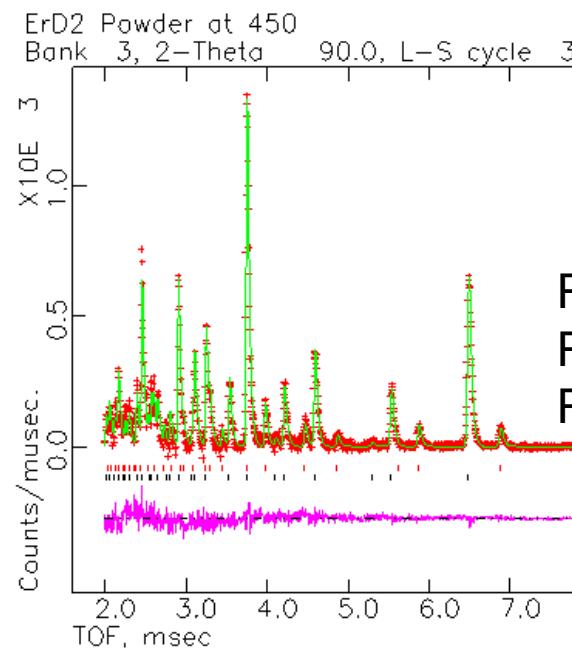
Formula: ErD<sub>tet(2.02)</sub>D<sub>oct(0.02)</sub>

$$\text{Vol} = 134.11 \text{ Å}^3$$

$$D_x = 8.48 \text{ g/cm}^3$$

Silicon 640c used  
as internal standard

D<sub>oct</sub> ~2%



# Neutron Powder Diffraction results - FY04

## ErD<sub>2</sub> powder (350 °C) loading

### Crystal structure data

<u>atom</u>	<u>x,y,z</u>	<u>occ.</u>	<u>B(Å<sup>2</sup>)</u>
Er	0,0,0	1.0	0.28(2)
D <sub>tet</sub>	1/4,1/4,1/4	1.01(1)	1.25(3)
D <sub>oct</sub>	1/2,1/2,1/2	0.10(1)	1.25

$$a = 5.1166(2) \text{ Å}$$

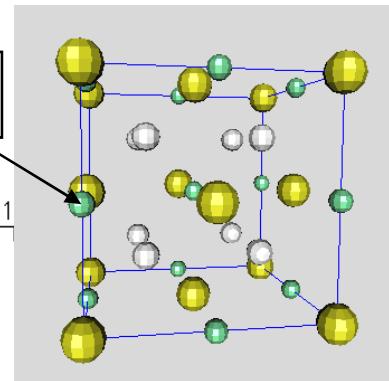
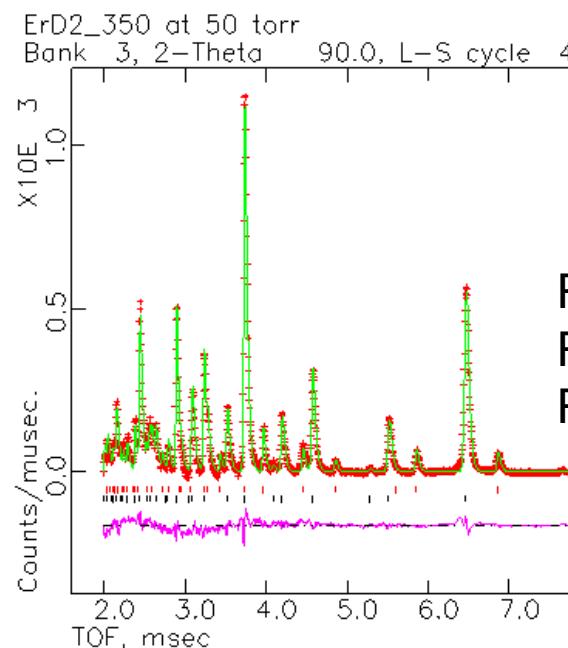
Formula: ErD<sub>tet(2.02)</sub>D<sub>oct(0.10)</sub>

$$\text{Vol} = 133.95 \text{ Å}^3$$

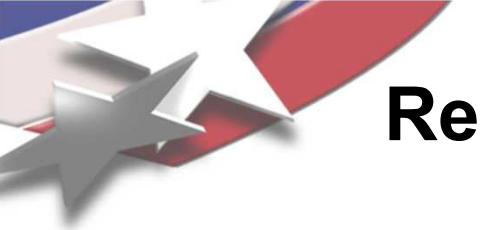
$$D_x = 8.51 \text{ g/cm}^3$$

Silicon 640c used  
as internal standard

D<sub>oct</sub> ~10%



XRD shows  $a = 5.116(1)$



# Results on ErD<sub>2</sub> powders – FY04

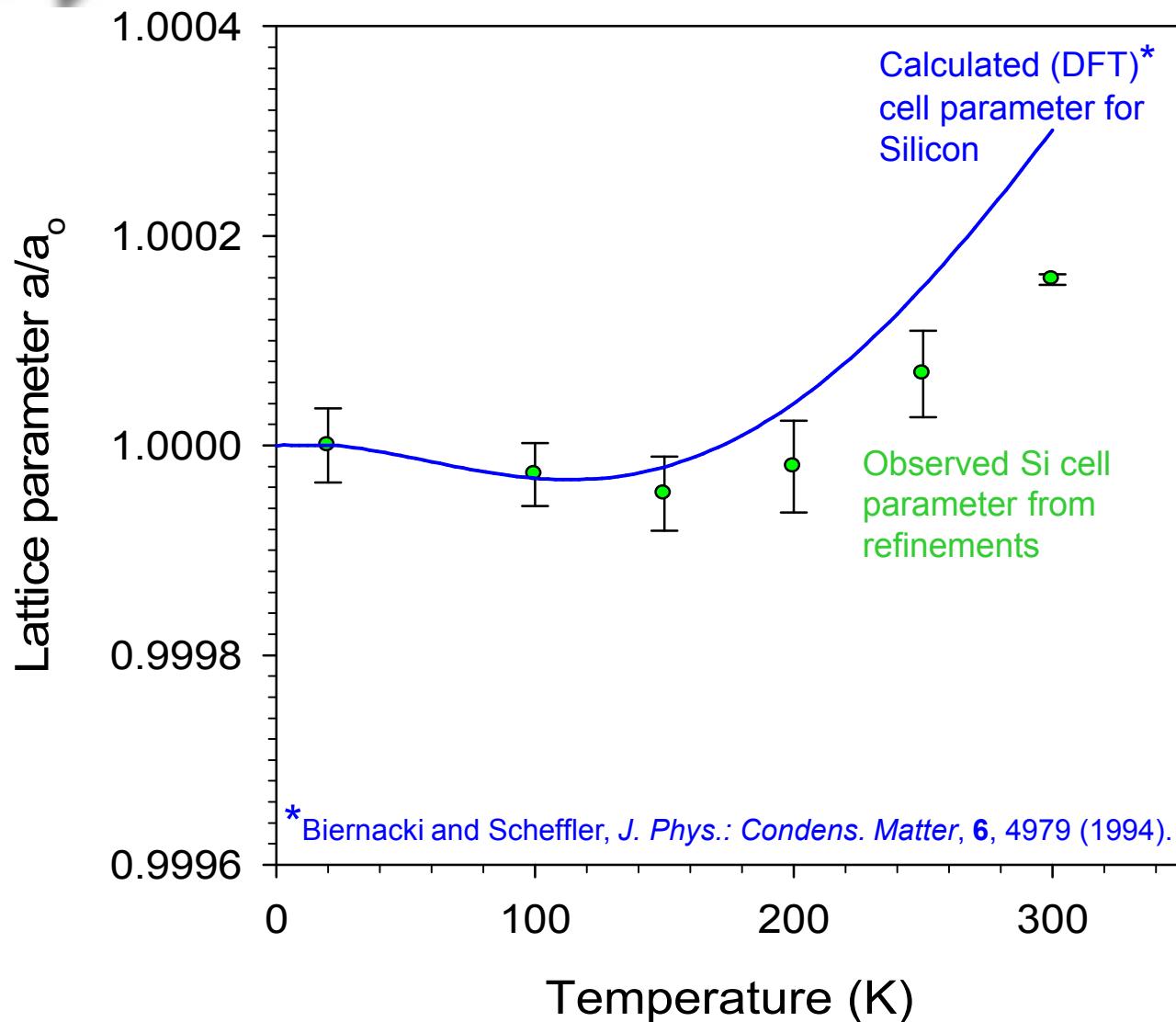
- At same pressure, lowering temperature (450 → 350 °C) encourages octahedral site occupancy.
- A corresponding shrinkage of the unit cell is observed with increased octahedral site occupancy.
- Tetrahedral site occupancy is essentially full for both structures.



# More recent ErD<sub>2</sub> powders – FY05

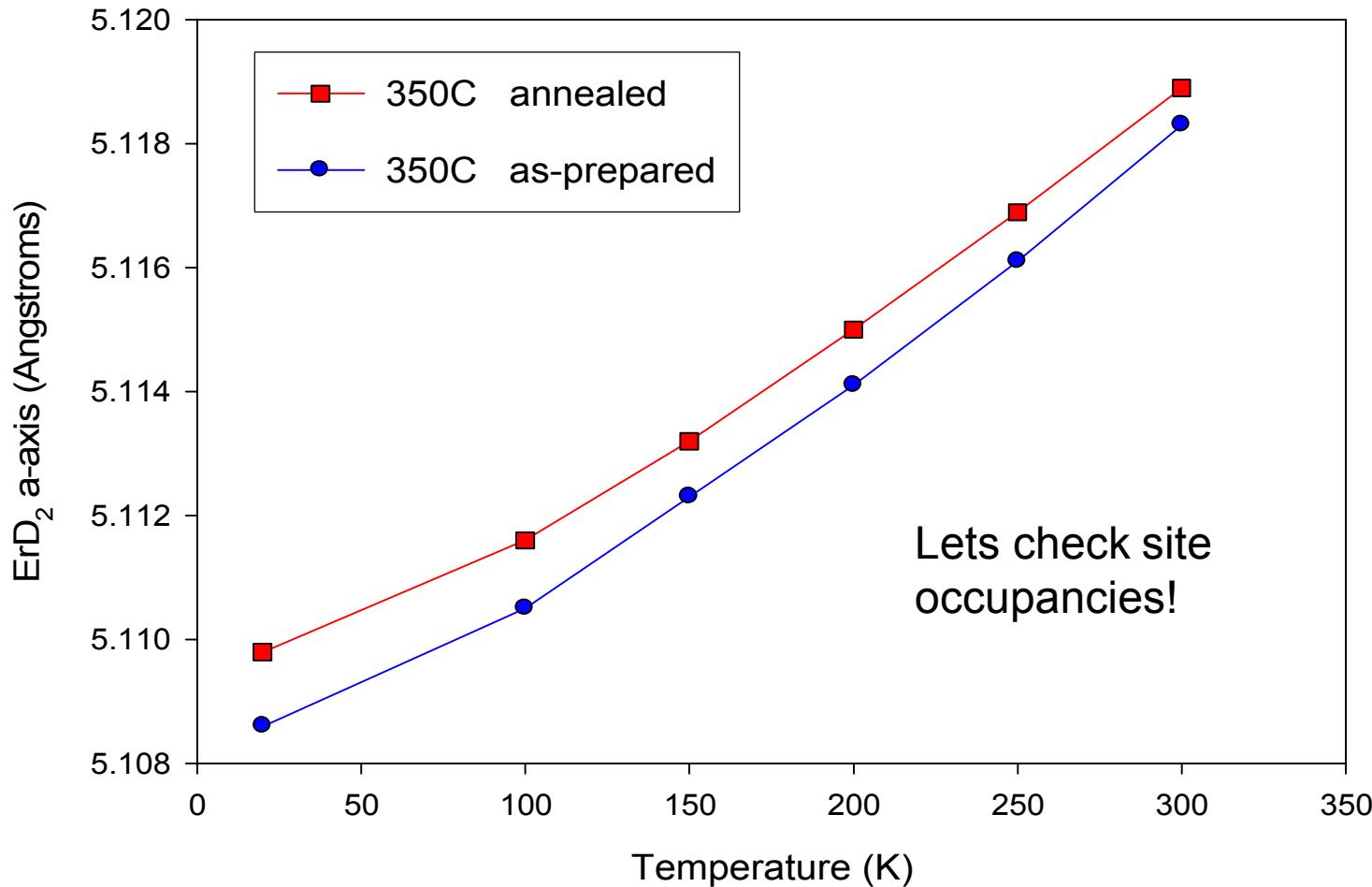
- Sample powders were prepared using same temperature and pressure (350 °C, 32 torr).
- One powder was analyzed “as-prepared”, while the other powder was annealed prior to neutron diffraction analysis.
- Powders were run as a function of temperature from 20 to 300 K.

Silicon lattice parameters were modeled as 2<sup>nd</sup> phase and confirm accuracy of observed ErD<sub>2</sub> lattice parameters

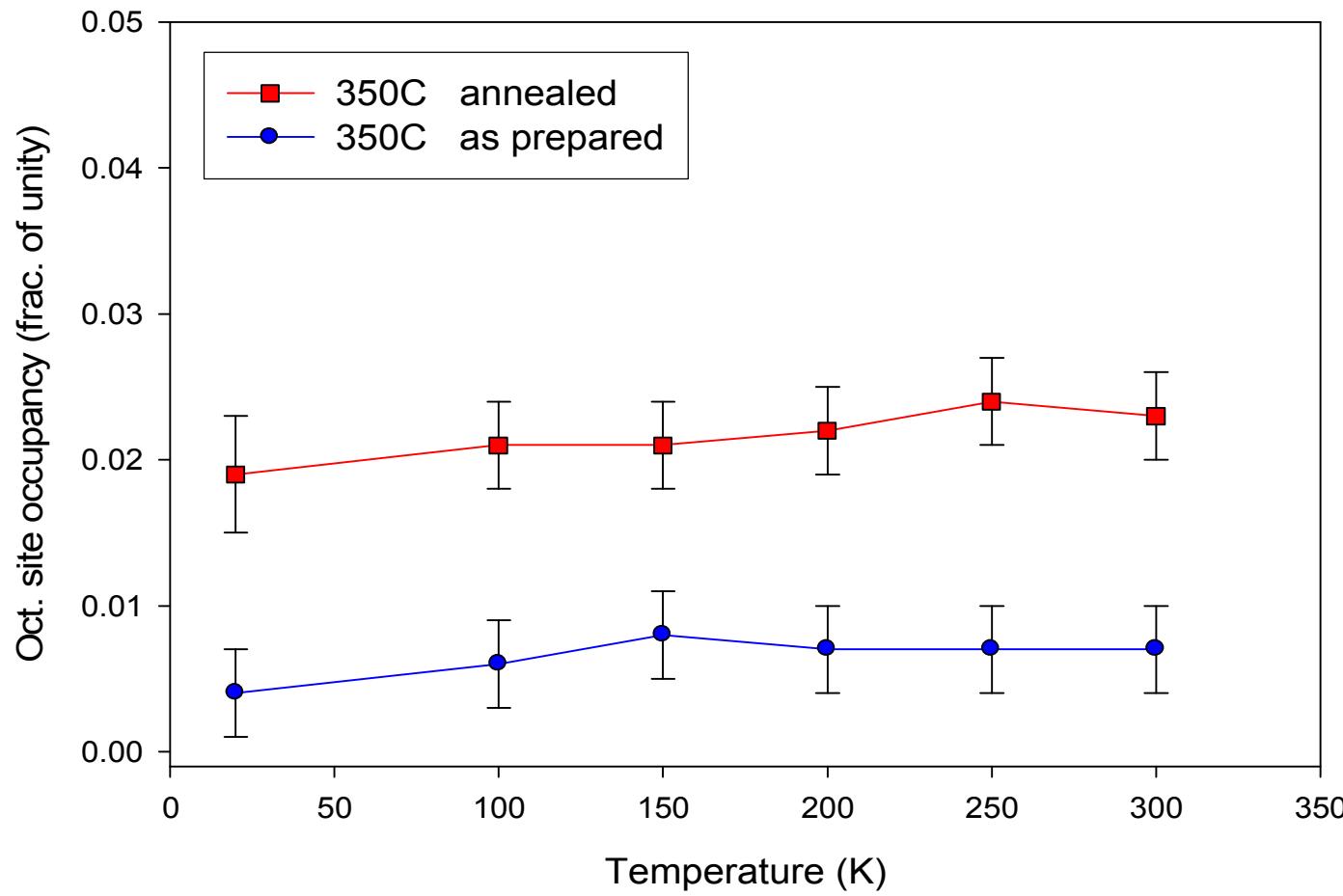


Temperature (K)	Si lattice parameter (Angstroms)
20	5.43033(19)
100	5.43018(16)
150	5.43008(19)
200	5.43022(24)
250	5.43070(22)
300	5.43119(05)

ErD<sub>2</sub> lattice parameter for annealed powder showed consistently larger unit cell as compared to as-prepared.



Annealed powder has higher occupancy (~2%) of octahedral site when compared with the as-prepared powder.





# What is happening here?

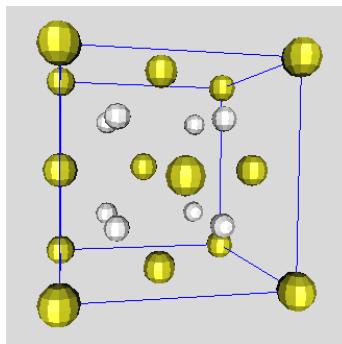
- These powders should be essentially the same. Annealing was only difference.
- Annealed sample shows larger unit cell *but also has a higher occupancy of the octahedral site.*
- FY04 analysis showed that the cell contracted with  $D_{\text{oct}}$  occupancy.



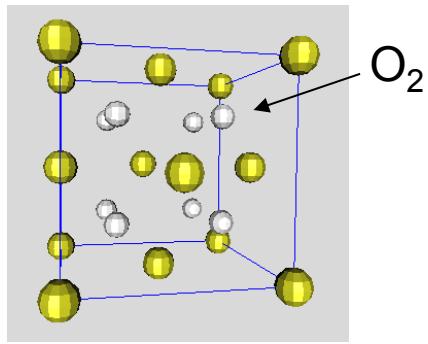
# Working theory regarding effect of annealing

- Additional annealing step allowed for oxygen solubility within the  $\text{ErD}_2$  lattice.

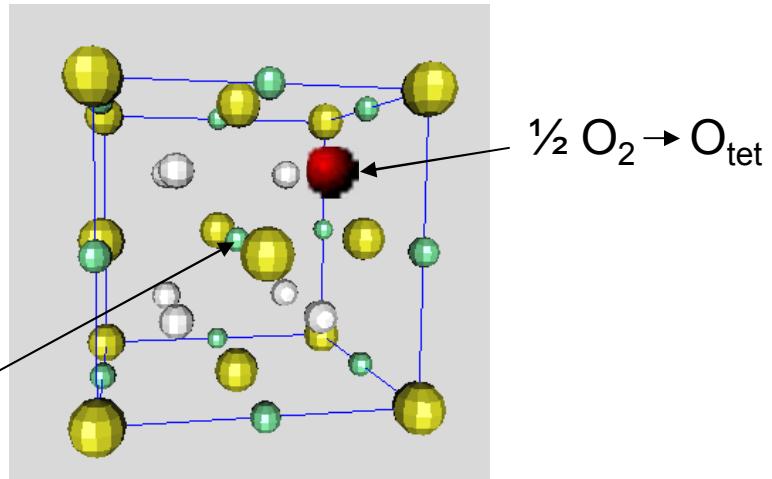
$\text{ErD}_2$  without octahedral D



Annealing



$D_{\text{tet}} \rightarrow D_{\text{oct}}$



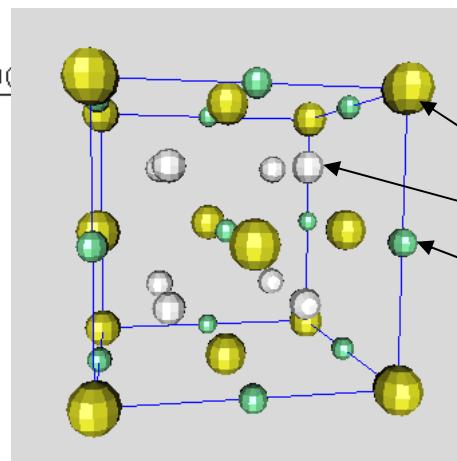
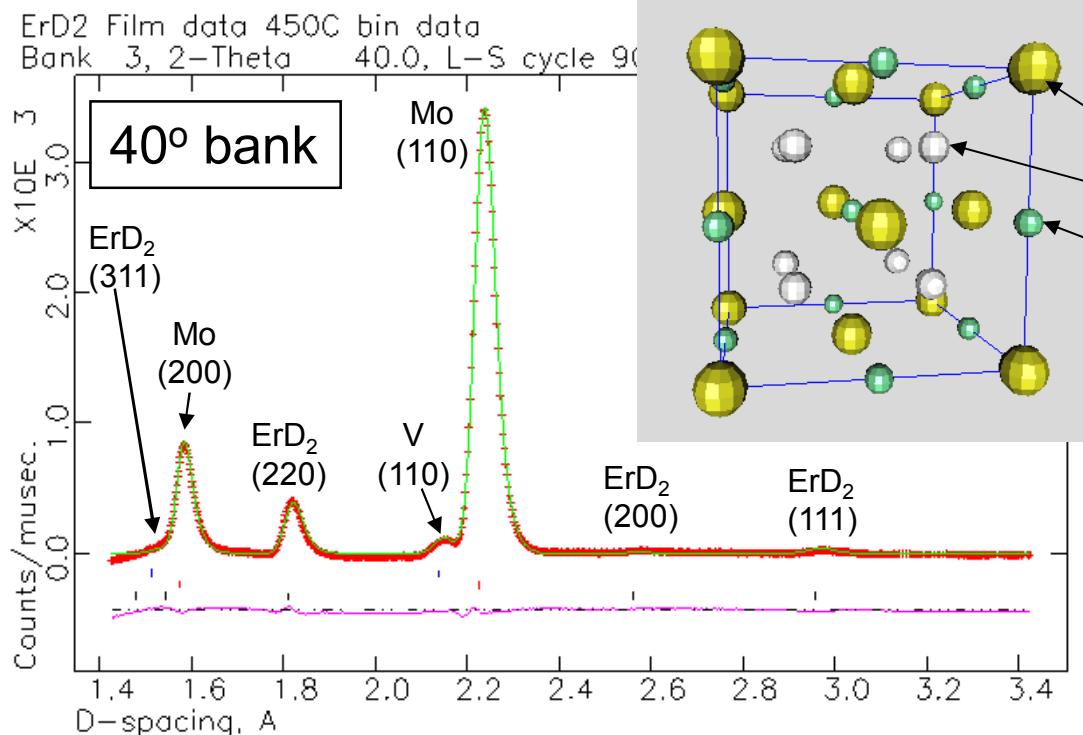
- Incorporation of oxygen defect at tetrahedral site expands lattice and drives tetrahedral deuterium to octahedral site.



# Summary

- **Presence of deuterium at octahedral site (~10% site occupancy) contracted the overall cell volume of the  $\text{ErD}_2$  fluorite lattice.**
- **The  $\text{ErD}_2$  lattice contracts uniformly with decreased temperature from 300 to 20 K.**
- **Annealed  $\text{ErD}_2$  powder showed significant fluorite unit-cell expansion and corresponding increase in octahedral site occupancy when compared against as-prepared sample.**
- **Annealing result may be explained by the addition of oxygen defects within the host lattice.**

Neutron scattering analysis of  $\text{ErD}_2$  thin films from FY05 run cycle (450C D-loading) shows similar result to that of annealed sample. Suggests oxygen defect presence in films.



Crystal data

Atom	occ.	$B_{\text{iso}}(\text{\AA}^2)$
Er	1.0	0.15
D <sub>t</sub>	1.04(2)	1.01
D <sub>o</sub>	0.15(3)	1.01
$a = 5.119(1) \text{\AA}$		
$\text{Vol} = 134.2 \text{\AA}^3$		
$R_{\text{wp}} = 0.0087$		
$R_{\text{p}} = 0.0073$		

# Decrease in $D_{\text{tet}}$ site occupancy is consistent with doping of oxygen for deuterium in annealed $\text{ErD}_2$

