



Hydrogen and Helium Isotopes in Materials Conference

SAND2007-0682C

February 6-7, 2007

“Neutron Diffraction of ErD_2 powders”

Mark A. Rodriguez and James F. Browning

*Sandia National Laboratories
Albuquerque, New Mexico 87185-1411*

&

Anna Llobet

*Los Alamos National Laboratory
Los Alamos, New Mexico 87545*

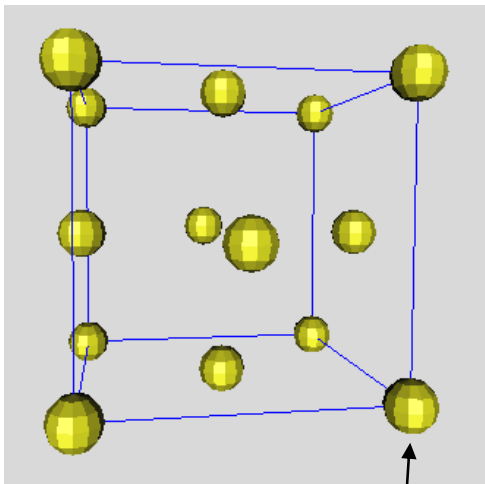


Structural Analysis of T-site occupancy in ErT_2 films is very challenging

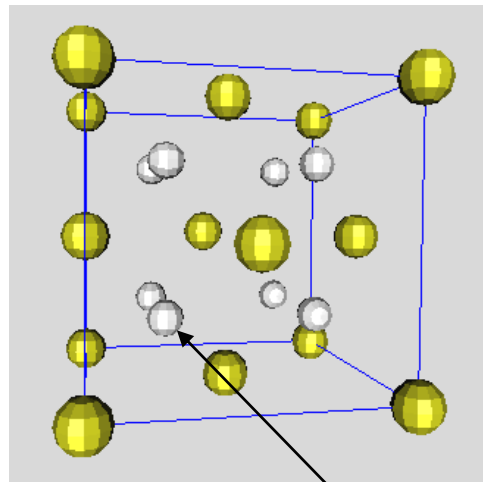
- Difficult determination of (H,D,T) in host lattice via XRD (i.e. only $1e^-$).
- (D,T) scatter well for neutron diffraction experiments.
- Strong texture of ErT_2 films complicates structural analysis.
- Loose powders are more random, providing straightforward analysis.
- Loose powder of ErT_2 is major contamination issue.
- Loose powders of ErD_2 chosen as surrogate powders for structural analysis of the fluorite lattice.
 - Gain insight into site occupancy without hazards of tritium

Fluorite structure of ErD_2

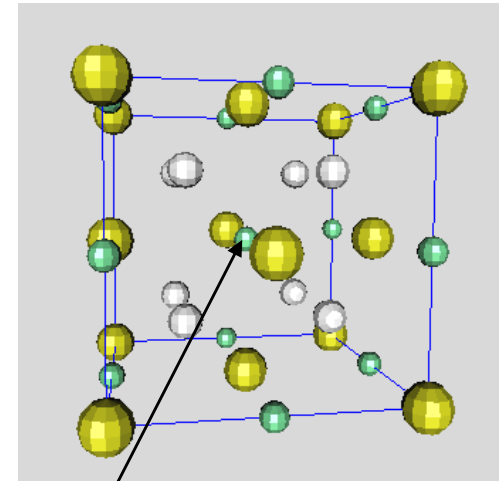
Where is the D going?



FCC lattice of Er



D occupies tetrahedral
($\frac{1}{4}, \frac{1}{4}, \frac{1}{4}$) sites



Octahedral ($\frac{1}{2}, \frac{1}{2}, \frac{1}{2}$)
sites could also contain D*

* T. J. Udovic, J.J. Rush, and I. S. Anderson, *Phys. Rev. B*, **50**, pp.7144 (1994)

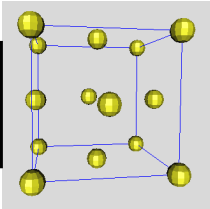


ErD₂ powder analysis

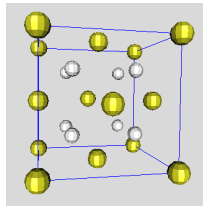
- Er metal D-loaded at various temperatures/pressures.
- ErD₂ powder specimens mixed with silicon powder standard.
- Powders placed in Vanadium canisters.
- Neutron diffraction data collected using HIPD at LANSCE/LANL.
- Structure refinements performed using GSAS software.
- Two run cycles (FY04 and FY05).

How D changes relative intensities (I_{rel}) for the ErD_2 fluorite-type structure

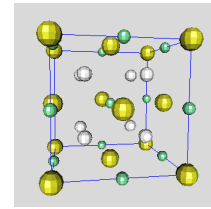
Model 1
Er only



Model 2
Er + D_{tet}



Model 3
Er + D_{tet}
+ 10% D_{oct}



XRD calculated patterns

| hkl | d (Å) | I_{rel} (%) Model 1 | I_{rel} (%) Model 2 | I_{rel} (%) Model 3 | Diff % 3-1 |
|-------|-------|-----------------------------|-----------------------------|-----------------------------|------------------|
| 111 | 2.954 | 100 | 100 | 100 | |
| 200 | 2.558 | 51 | 49 | 49 | -2 |
| 220 | 1.808 | 35 | 36 | 36 | +1 |
| 311 | 1.543 | 40 | 40 | 40 | |
| 222 | 1.477 | 11 | 11 | 11 | |
| 400 | 1.279 | 5 | 5 | 5 | |
| 331 | 1.174 | 15 | 15 | 15 | |
| 420 | 1.144 | 13 | 13 | 13 | |
| 422 | 1.044 | 10 | 10 | 10 | |

Neutron calculated patterns

| hkl | d (Å) | I_{rel} (%) Model 1 | I_{rel} (%) Model 2 | I_{rel} (%) Model 3 | Diff. % 3-1 |
|-------|-------|-----------------------------|-----------------------------|-----------------------------|-------------------|
| 111 | 2.954 | 100 | 26 | 21 | -79 |
| 200 | 2.558 | 56 | 6 | 5 | -51 |
| 220 | 1.808 | 55 | 100 | 100 | +45 |
| 311 | 1.543 | 78 | 20 | 16 | -62 |
| 222 | 1.477 | 24 | 3 | 2 | -22 |
| 400 | 1.279 | 13 | 24 | 24 | +11 |
| 331 | 1.174 | 45 | 12 | 9 | -36 |
| 420 | 1.144 | 43 | 5 | 4 | -39 |
| 422 | 1.044 | 36 | 67 | 67 | +31 |

Neutron Powder Diffraction results - FY04

ErD₂ powder (450 °C) loading

Crystal structure data

| atom | x,y,z | occ. | B(Å ²) |
|------------------|-------------|---------|--------------------|
| Er | 0,0,0 | 1.0 | 0.15(2) |
| D _{tet} | 1/4,1/4,1/4 | 1.00(1) | 1.01(2) |
| D _{oct} | 1/2,1/2,1/2 | 0.02(1) | 1.01 |

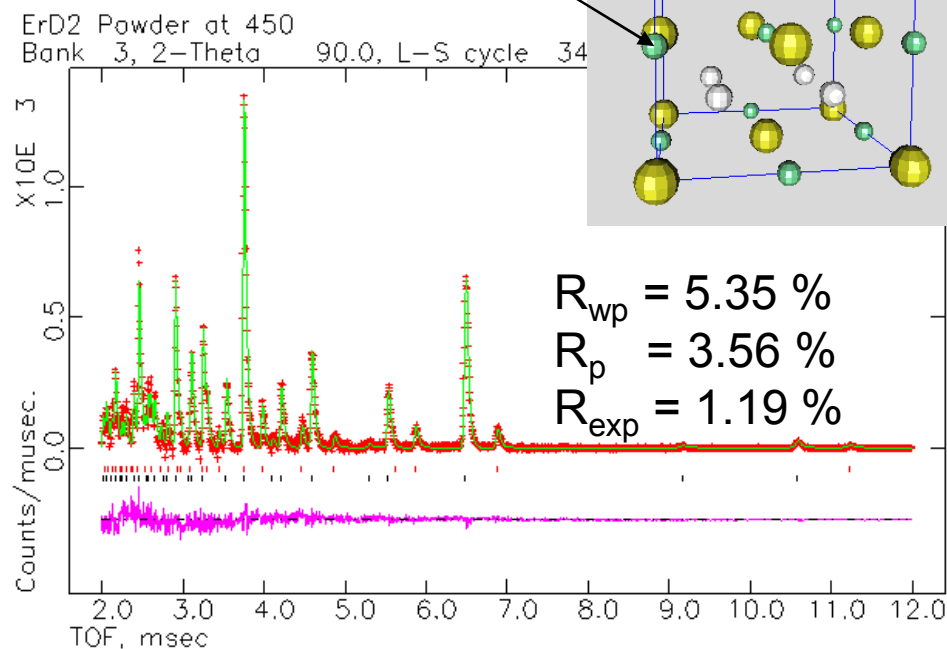
$$a = 5.1187(2) \text{ \AA}$$

$$\text{Formula: ErD}_{\text{tet}(2.02)}\text{D}_{\text{oct}(0.02)}$$

$$\text{Vol} = 134.11 \text{ \AA}^3$$

$$D_x = 8.48 \text{ g/cm}^3$$

D_{oct} ~2%



Silicon 640c used
as internal standard

Neutron Powder Diffraction results - FY04

ErD₂ powder (350 °C) loading

Crystal structure data

| atom | x,y,z | occ. | B(Å ²) |
|------------------|-------------|---------|--------------------|
| Er | 0,0,0 | 1.0 | 0.28(2) |
| D _{tet} | 1/4,1/4,1/4 | 1.01(1) | 1.25(3) |
| D _{oct} | 1/2,1/2,1/2 | 0.10(1) | 1.25 |

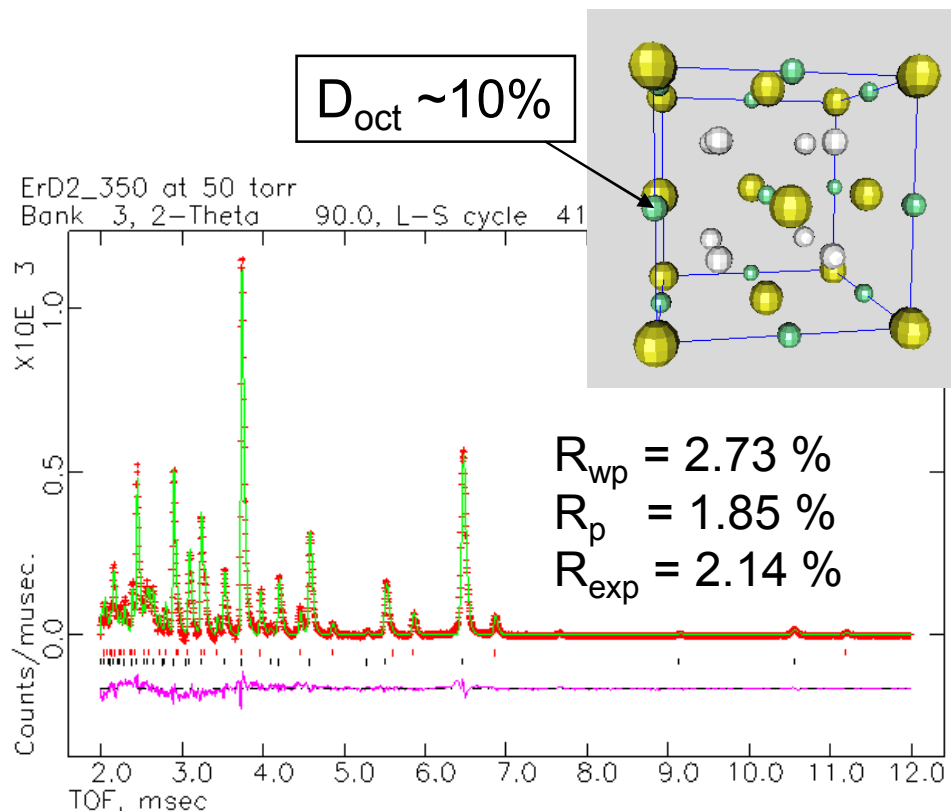
$a = 5.1166(2) \text{ \AA}$

Formula: $\text{ErD}_{\text{tet}(2.02)}\text{D}_{\text{oct}(0.10)}$

$\text{Vol} = 133.95 \text{ \AA}^3$

$D_x = 8.51 \text{ g/cm}^3$

Silicon 640c used
as internal standard



XRD shows $a = 5.116(1)$



Results on ErD_2 powders – FY04

- At same pressure, lowering temperature ($450 \rightarrow 350$ °C) encourages octahedral site occupancy.
- A corresponding shrinkage of the unit cell is observed with increased octahedral site occupancy.
- Tetrahedral site occupancy is essentially full for both structures.

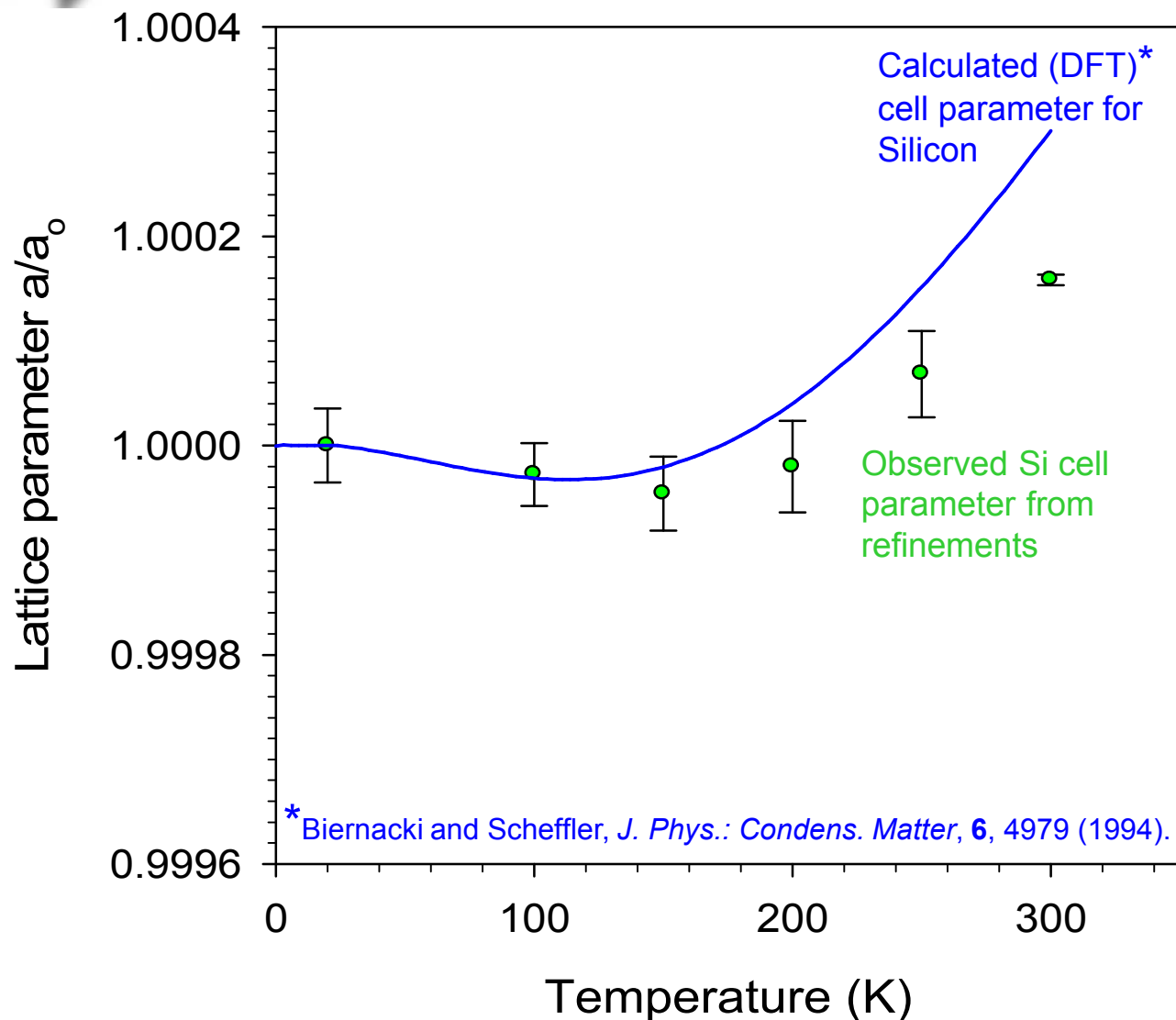


More recent ErD₂ powders – FY05

- Sample powders were prepared using same temperature and pressure (350 °C, 32 torr).
- One powder was analyzed “as-prepared”, while the other powder was annealed prior to neutron diffraction analysis.
- Powders were run as a function of temperature from 20 to 300 K.

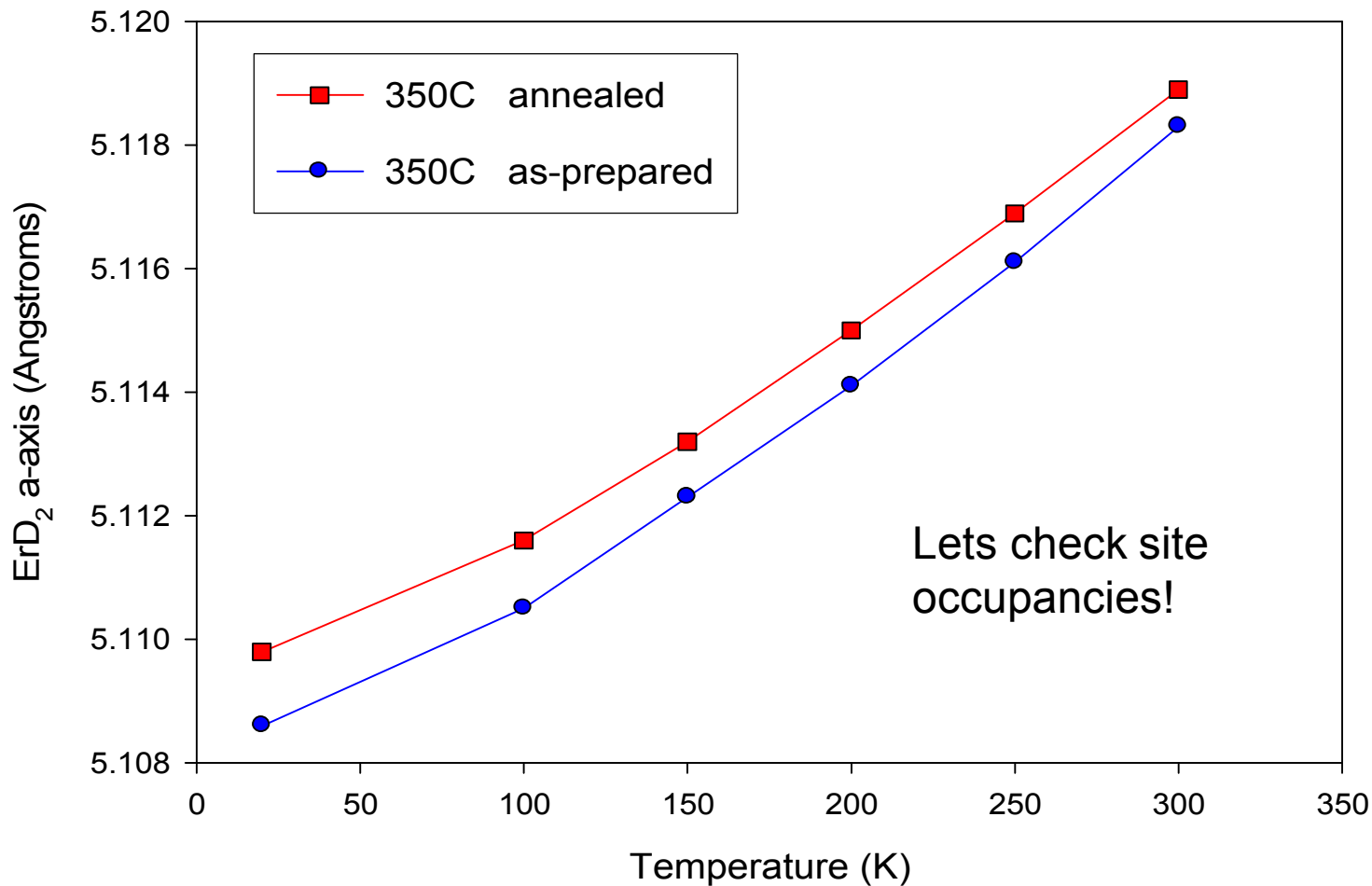



Silicon lattice parameters were modeled as 2nd phase and confirm accuracy of observed ErD₂ lattice parameters



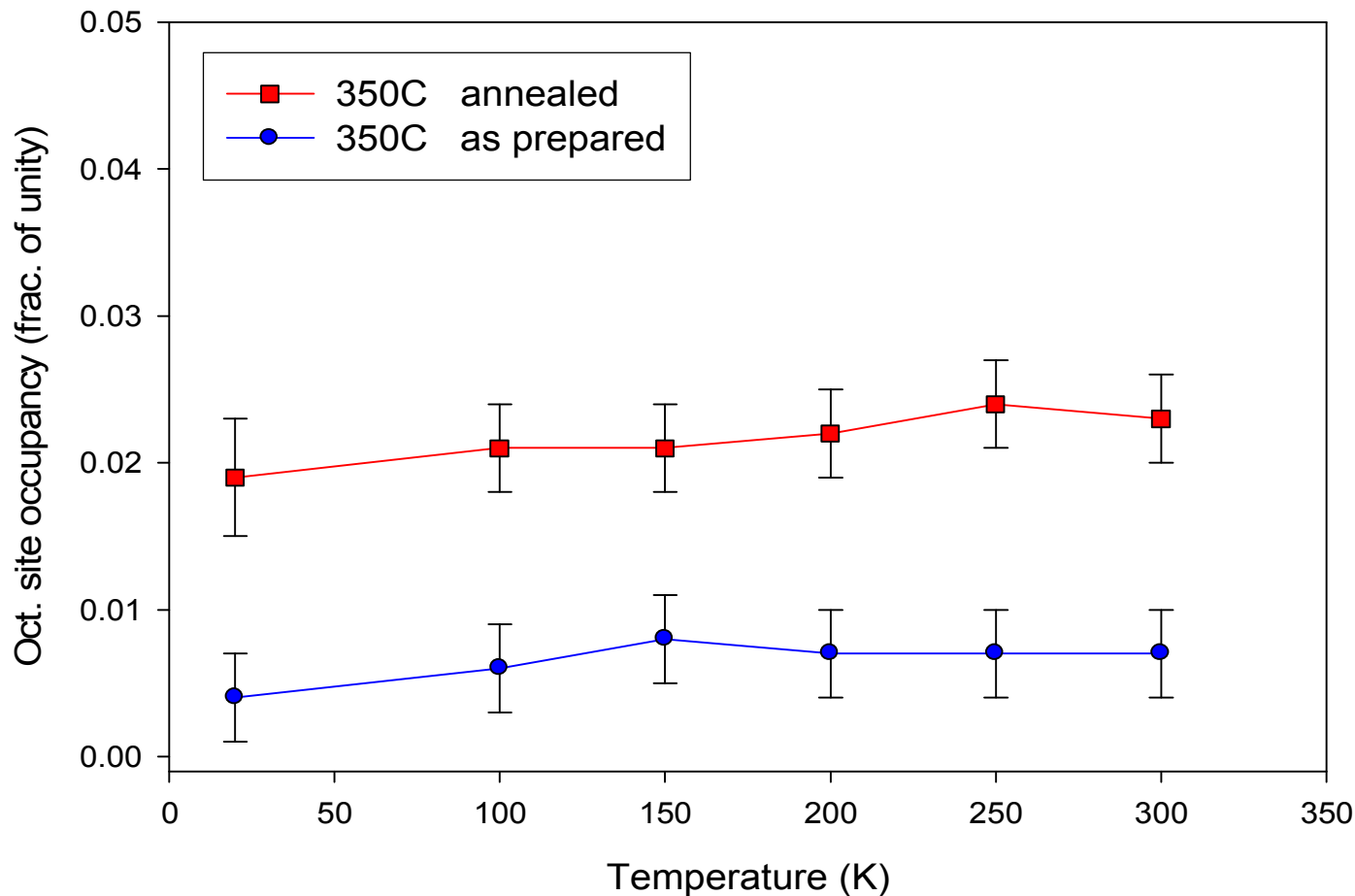
| Temperature (K) | Si lattice parameter (Angstroms) |
|-----------------|----------------------------------|
| 20 | 5.43033(19) |
| 100 | 5.43018(16) |
| 150 | 5.43008(19) |
| 200 | 5.43022(24) |
| 250 | 5.43070(22) |
| 300 | 5.43119(05) |

ErD₂ lattice parameter for annealed powder showed consistently larger unit cell as compared to as-prepared.





Annealed powder has higher occupancy (~2%) of octahedral site when compared with the as-prepared powder.



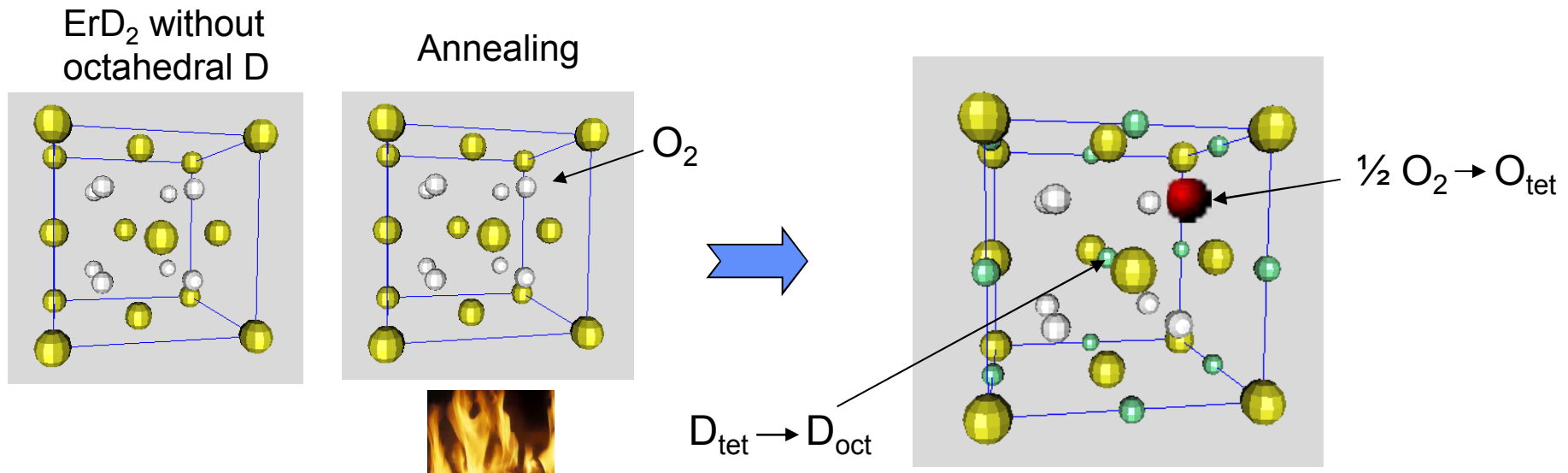


What is happening here?

- These powders should be essentially the same. Annealing was only difference.
- Annealed sample shows larger unit cell *but also has a higher occupancy of the octahedral site.*
- FY04 analysis showed that the cell contracted with D_{oct} occupancy.

Working theory regarding effect of annealing

- Additional annealing step allowed for oxygen solubility within the ErD_2 lattice.



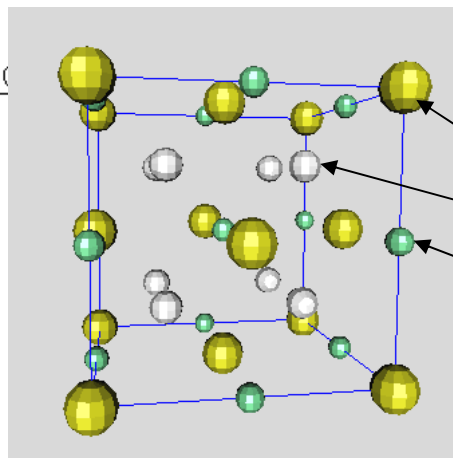
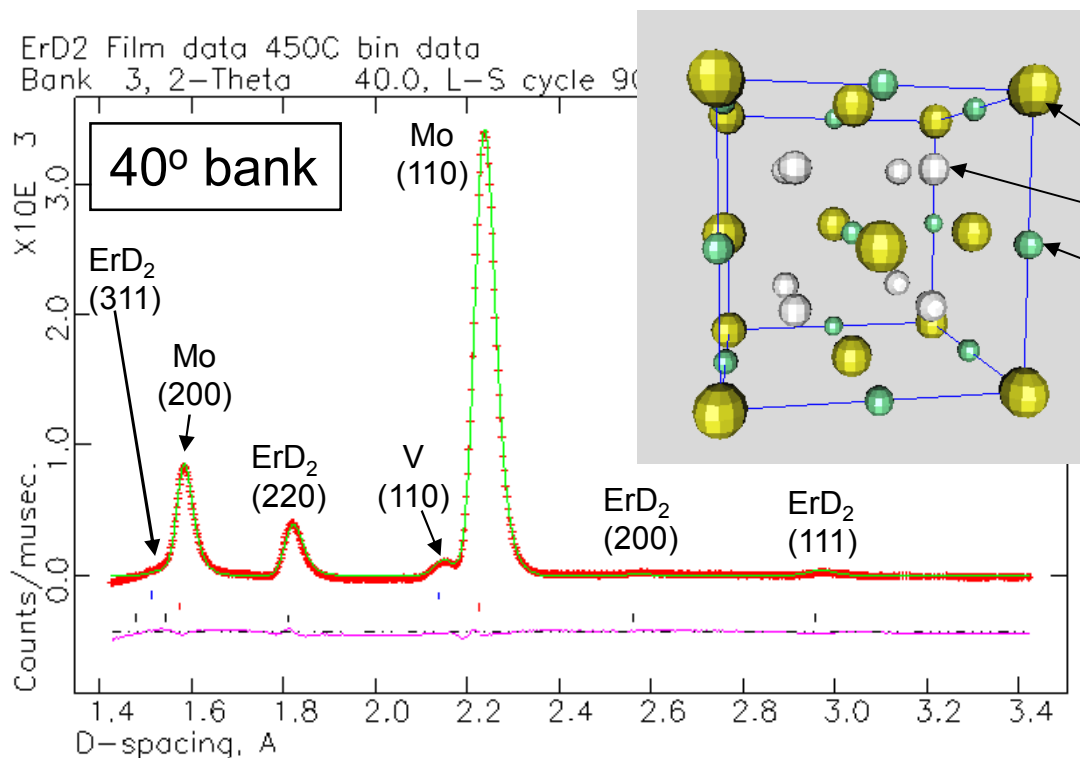
- Incorporation of oxygen defect at tetrahedral site expands lattice and drives tetrahedral deuterium to octahedral site.



Summary

- **Presence of deuterium at octahedral site (~10% site occupancy) contracted the overall cell volume of the ErD_2 fluorite lattice.**
- **The ErD_2 lattice contracts uniformly with decreased temperature from 300 to 20 K.**
- **Annealed ErD_2 powder showed significant fluorite unit-cell expansion and corresponding increase in octahedral site occupancy when compared against as-prepared sample.**
- **Annealing result may be explained by the addition of oxygen defects within the host lattice.**

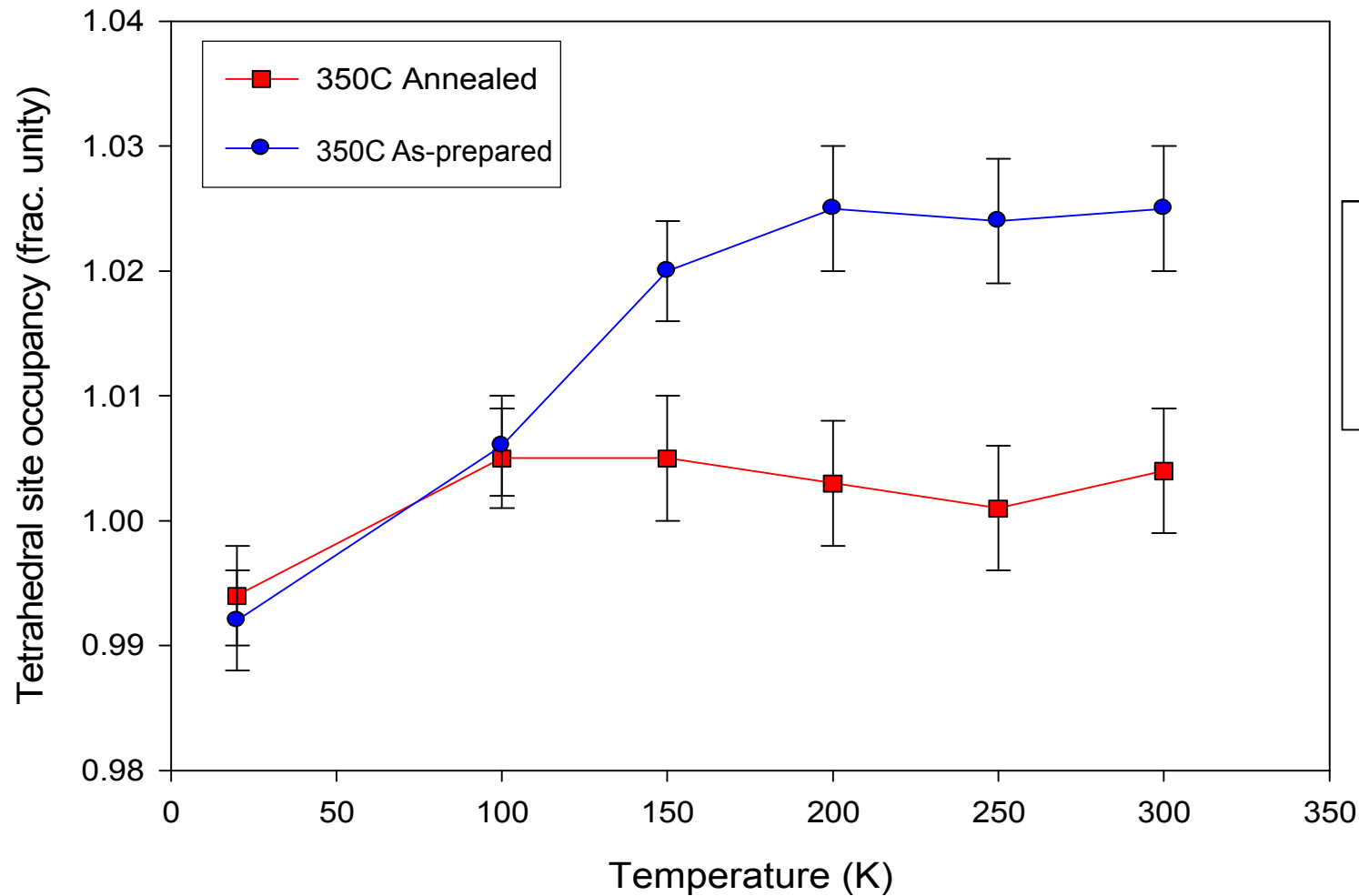
Neutron scattering analysis of ErD_2 thin films from FY05 run cycle (450C D-loading) shows similar result to that of annealed sample. Suggests oxygen defect presence in films.



Crystal data

| Atom | occ. | $B_{iso} (\text{\AA}^2)$ |
|------------------------------------|---------|--------------------------|
| Er | 1.0 | 0.15 |
| D _t | 1.04(2) | 1.01 |
| D _o | 0.15(3) | 1.01 |
| $a = 5.119(1) \text{ \AA}$ | | |
| $\text{Vol} = 134.2 \text{ \AA}^3$ | | |
| $R_{wp} = 0.0087$ | | |
| $R_p = 0.0073$ | | |

Decrease in D_{tet} site occupancy is consistent with doping of oxygen for deuterium in annealed ErD_2



| Atom name | n-Scattering length (fm) |
|-----------|--------------------------|
| Deuterium | 6.67 |
| Oxygen | 5.80 |
| Erbium | 7.79 |