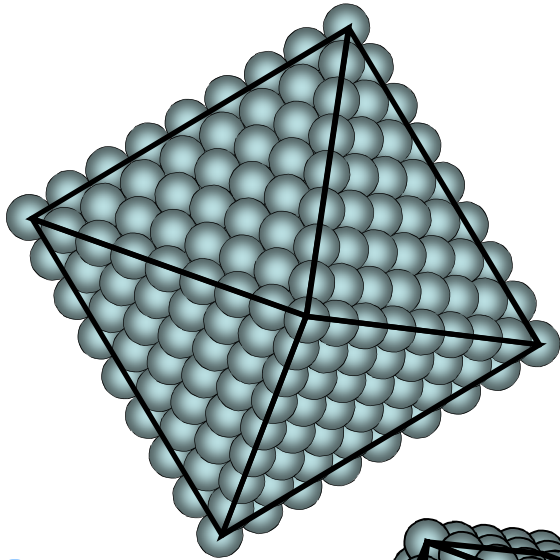
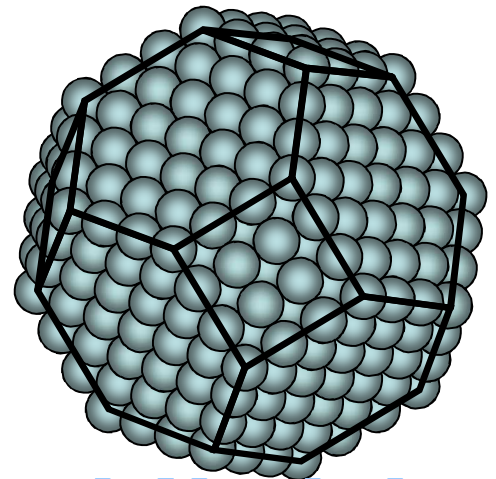


# Pb Nanoprecipitates in Al: Magic-Shape Effects Due to Elastic Strain

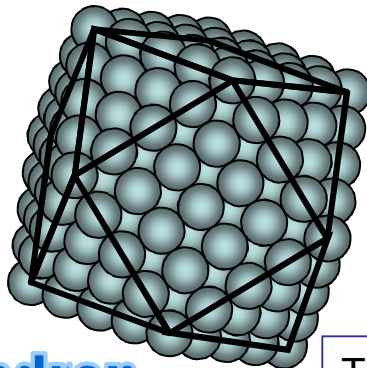
John Hamilton, Francois Leonard,  
and Ulrich Dahmen



**Octahedron**  
 **$C/A=1.73$**



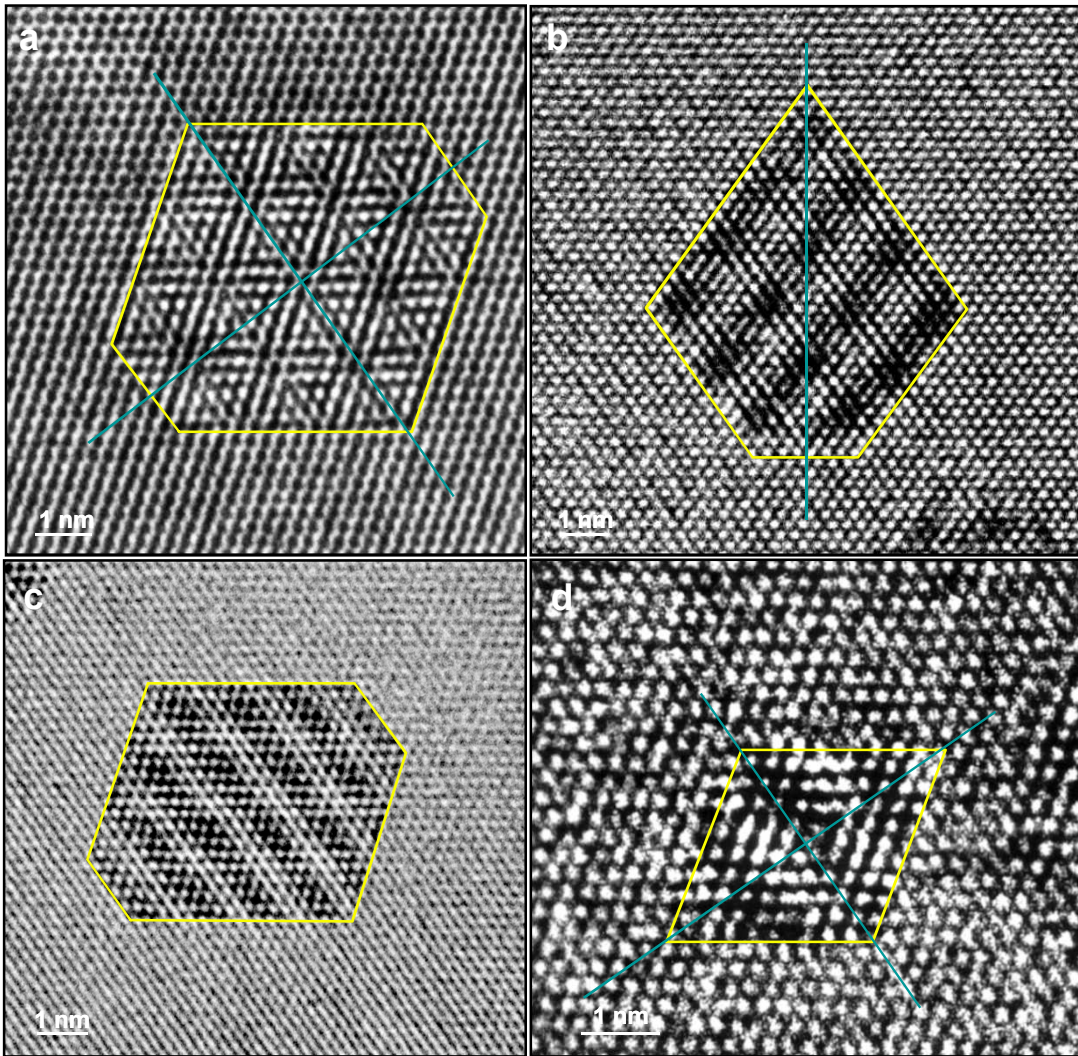
**Tetrakaidecahedron**  
 **$C/A=1.16$**



**Cuboctahedron**  
 **$C/A=0.86$**

These figures illustrate three regular polyhedra with  $O_h$  symmetry, spanning the range of aspect ratios,  $C/A$ , observed for Pb precipitates in Al.

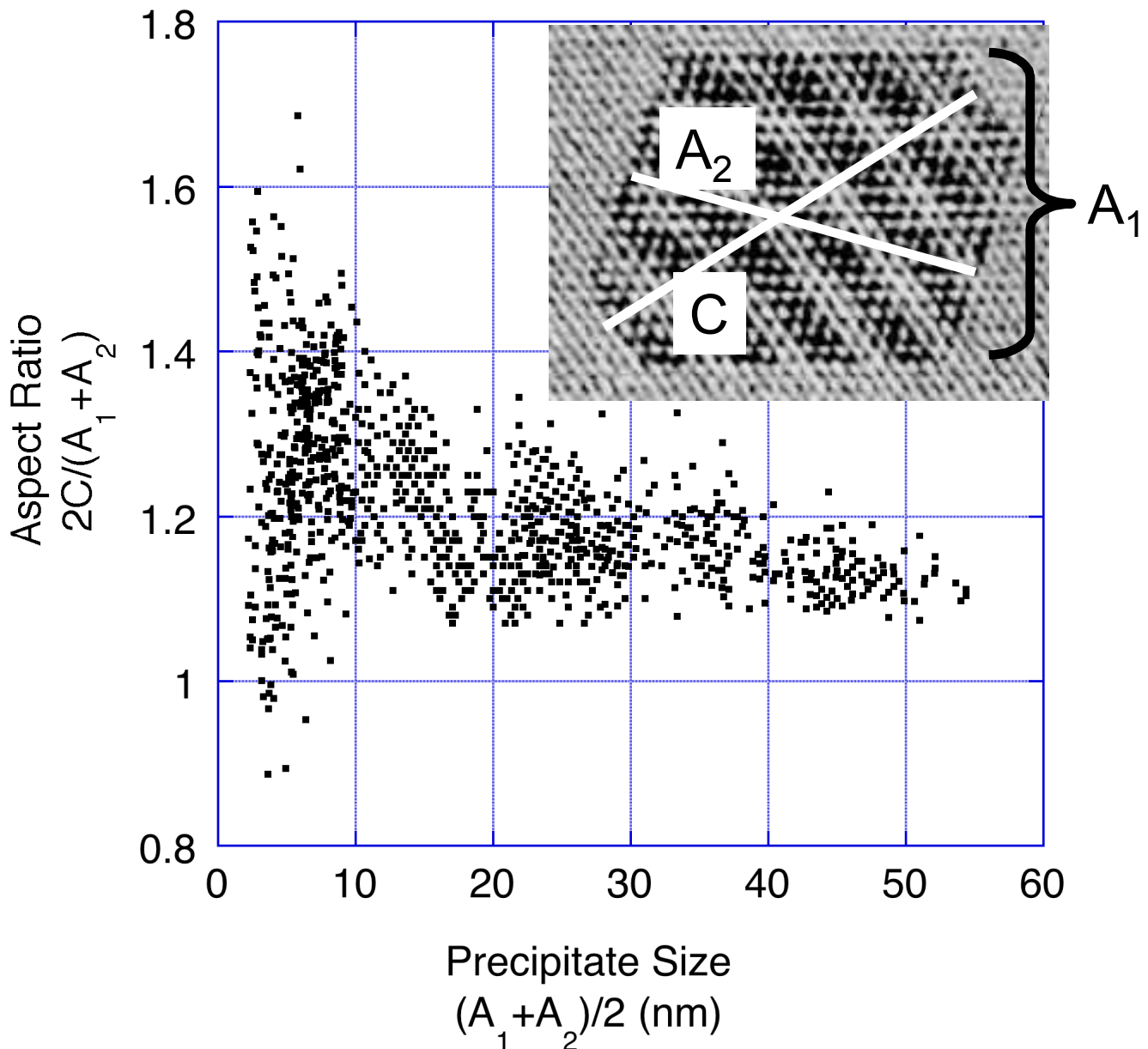
# TEM Micrographs



- These micrographs suggest the variety of shapes observed experimentally for small Pb precipitates in Al.
- (a) shows a precipitate which is approximately tetrakaidecahedral as predicted by the Wulff construction.
- (b) shows an asymmetrical precipitate (bottom vertex truncated).
- (c) shows an asymmetrical precipitate (unequal distances between pairs of  $\{111\}$  type facets).
- (d) shows a very small octahedral precipitate.

**Our Goal is to explain these size-dependent shape effects.**

# Aspect Ratio from TEM

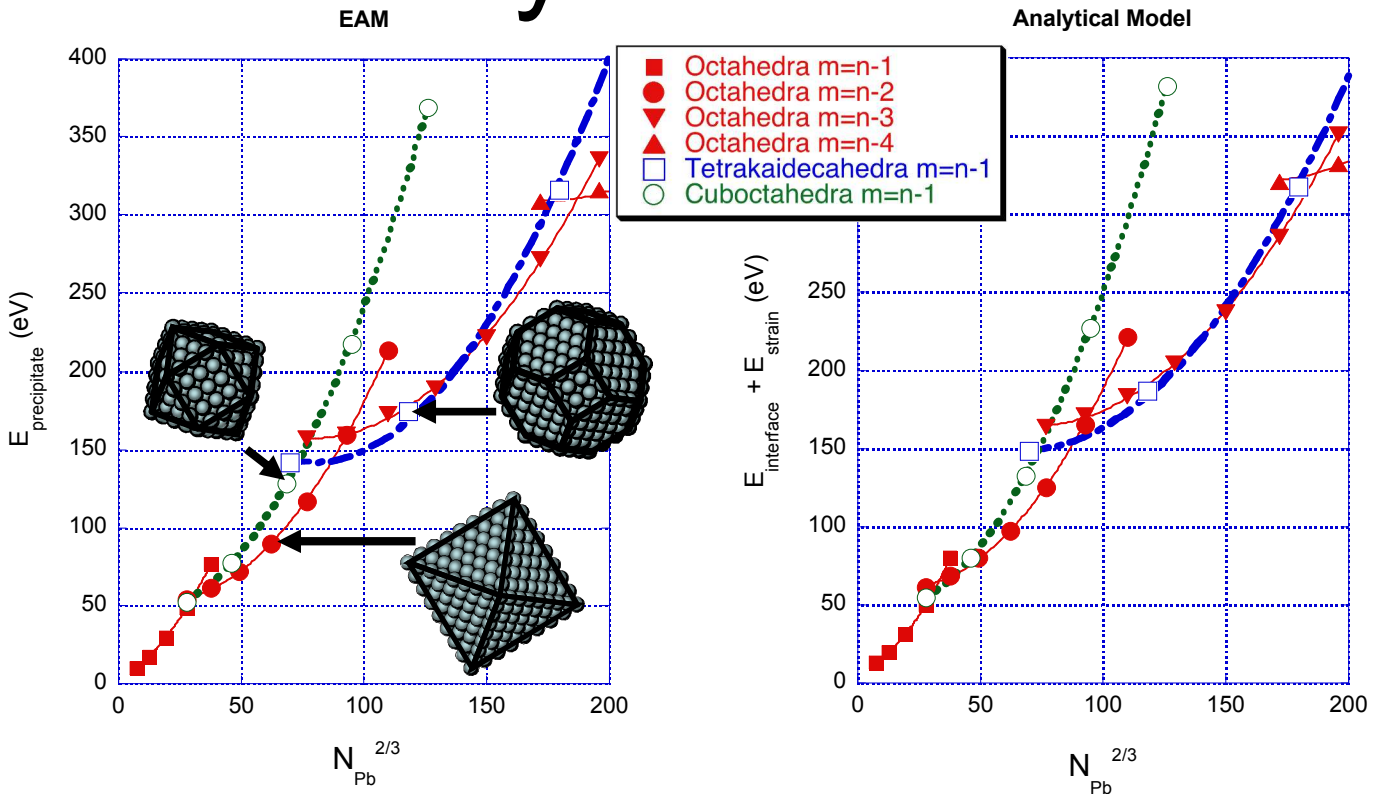


- For large precipitates, the Wulff construction predicts an aspect ratio of 1.16, in excellent agreement with experiment.
- For small precipitates, a variety of shapes are observed and the range of aspect ratios increases dramatically.

**Our Goal is to explain this size-dependent shape effect.**



# We Used EAM and an Analytical Model

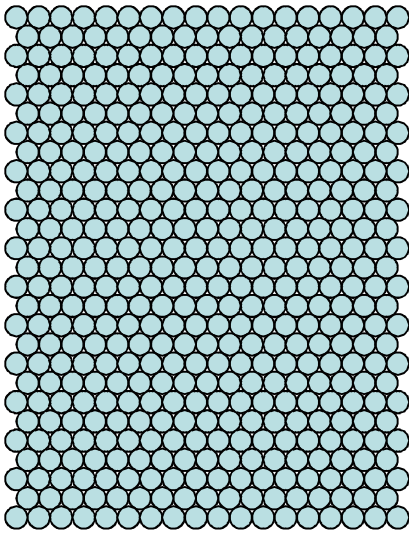


- Atomistic calculation (EAM) gives precipitate energy as a function of shape and size.
- Analytic Model includes strain energy and interface energy, **but NOT edge energy**.
- EAM and Analytical Model agree proving that edge energy does not determine shape.
- The EAM and Analytical calculations are the subject of next several pages.

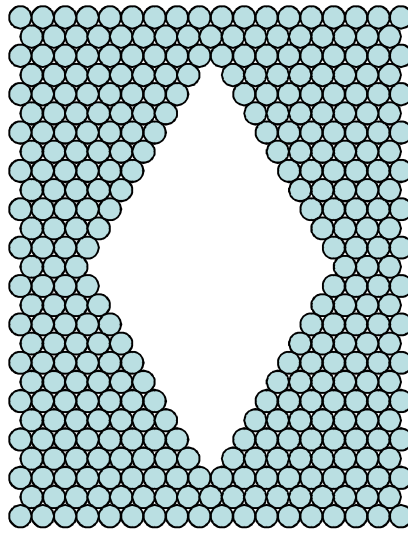
**Edge Energy Does Not Explain Shape Effects!**  
**What Does?**



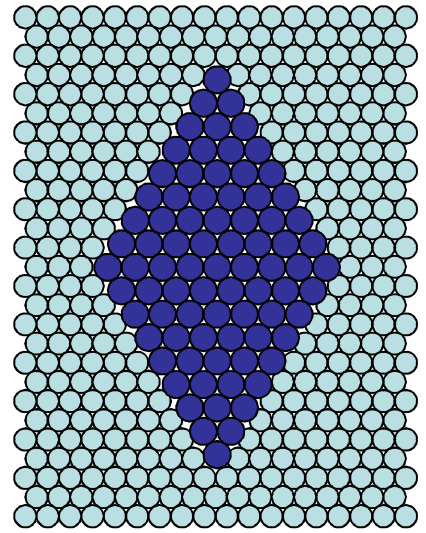
# EAM Calculation



Start with Al  
slab

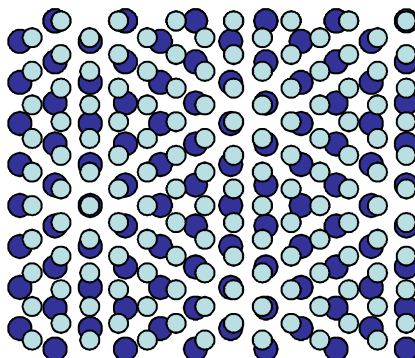


Remove Al octahedron  
 $n_{\text{oct}}=11$  atoms on edge



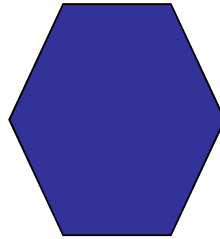
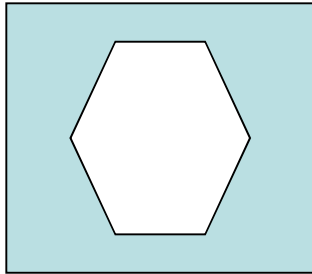
Insert Pb octahedron  
 $m_{\text{oct}}=9$  atoms on edge

- The lattice constants are  $a_{\text{Al}}=4.05\text{\AA}$  and  $a_{\text{Pb}}=4.95\text{\AA}$
- Since  $a_{\text{Pb}}/a_{\text{Al}}=11/9$ , a regular polyhedron (in this case an octahedron) with  $m=9$  Pb atoms on an edge fits with zero strain in the void made by removing the same regular polyhedron with  $n=11$  Al atoms on an edge.
- After relaxing atomic coordinates, EAM calculation gives  $E_{\text{total}}$  for Pb precipitate in Al slab. The total energy of the precipitate is  $E_{\text{precipitate}}=E_{\text{total}}-N_{\text{Al}} E_{\text{Al}}-N_{\text{Pb}} E_{\text{Pb}}$ .
- We performed this calculation for a range of sizes and for octahedra, tetrakaidecahedra, and cuboctahedra.
- Due to large size difference, Al/Pb interfaces are not coherent.



# Continuum Model

Al matrix  
with void



Pb precipitate  
with same shape

- We neglect edge energy.

$$E_{\text{precipitate}} = E_{\text{strain}} + E_{\text{interface}}$$

- We consider case with homogeneous strain (i.e. similar shape for void and inserted precipitate).

$$E_{\text{strain}} = \frac{Y a_{\text{Pb}}^3 \eta^2}{4(1-\nu)} N_{\text{Pb}} \quad \eta = 1 - \left( \frac{N_{\text{Pb}}}{N_{\text{Al}}} \right)^{1/3} \frac{a_{\text{Pb}}}{a_{\text{Al}}}$$

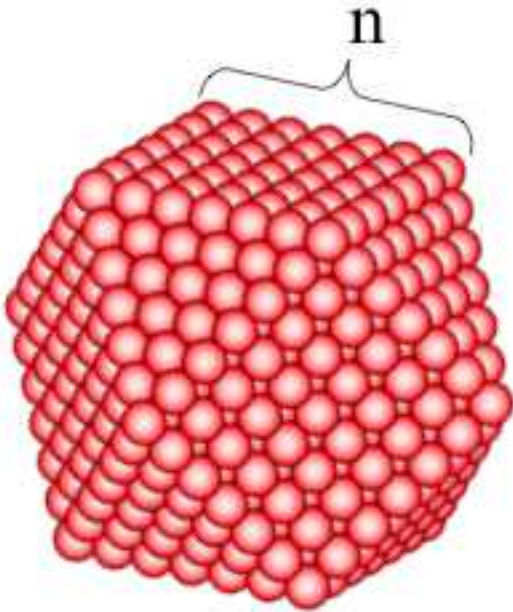
- Interface energy

$$E_{\text{interface}} = A_{111} \gamma_{111} + A_{100} \gamma_{100}$$

QuickTime™ and a  
TIFF (Uncompressed) decompressor  
are needed to see this picture.

**At nanoscale, facet  
lengths, facet areas,  
and edge energies  
are ill-defined!**

# Defining Edge Lengths\*



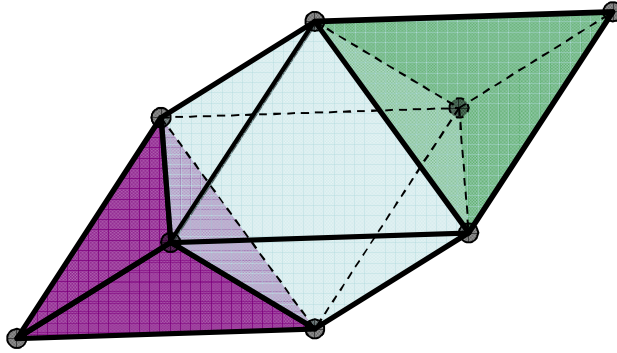
- $n$ , the number of atoms on an edge, is well-defined.
- $s$ , the precise edge length in length units, say Å, is not well-defined.
- There is a fundamental problem in defining edge lengths, facet areas, and in calculating edge energies.
- Problem is related to concept of Gibbs dividing surfaces which addresses ambiguity in definition of surface position.
- This problem is solved by using Gibbs equimolar surfaces to define the facets. The facet intersections define edges precisely.
- For a cuboctahedron,

$$s = \sqrt[3]{n^3 - \frac{3n^2}{2} + \frac{11n}{10} - \frac{3}{10}} \left( \frac{a}{\sqrt{2}} \right) \approx \left[ n - \frac{1}{2} + \frac{7}{60n} \right] \left( \frac{a}{\sqrt{2}} \right)$$

\* “Edge energies: Atomistic calculations of a continuum quantity”,  
J.C. Hamilton, Phys. Rev. B **73**, 125447 (2006).

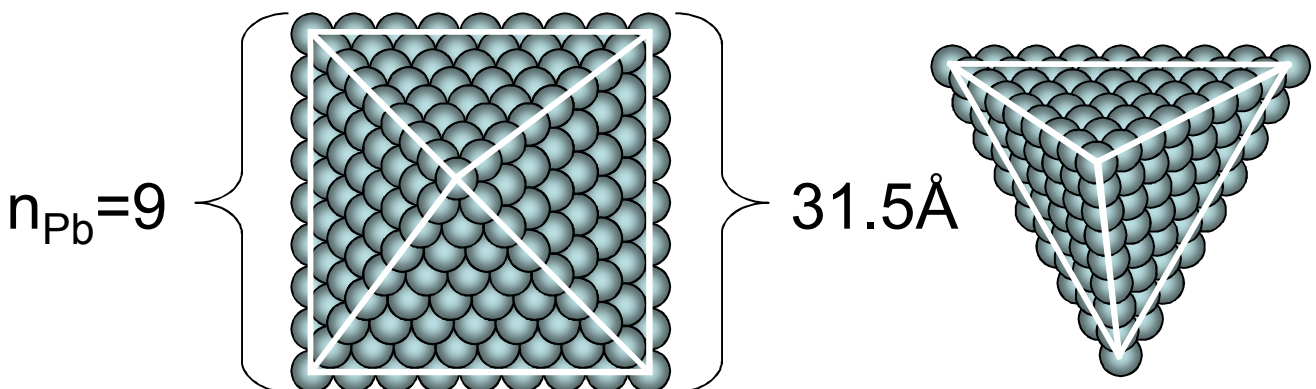


# “Magic Shapes” are Shapes with Zero Strain

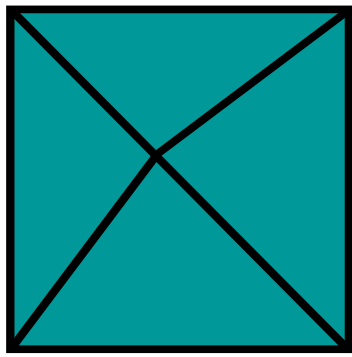


- Rhombohedral unit cell of fcc lattice can be constructed from two square pyramids and two tetrahedra.
- Consequently strain-free Pb precipitate in Al, must be constructed from “building blocks” in the shape of tetrahedra and square pyramids.
- For zero strain, building blocks must be square pyramids and tetrahedra with 9 Pb atoms on an edge.

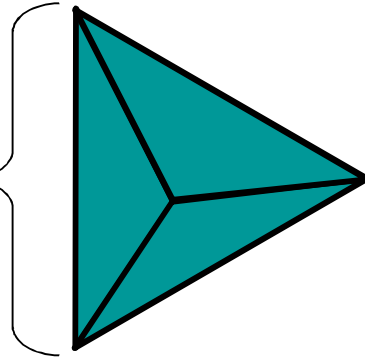
## Building Blocks for Zero Strain Precipitates



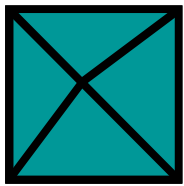
# Include “Magic-Shapes” with Small Homogeneous Strains



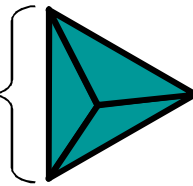
31.5Å



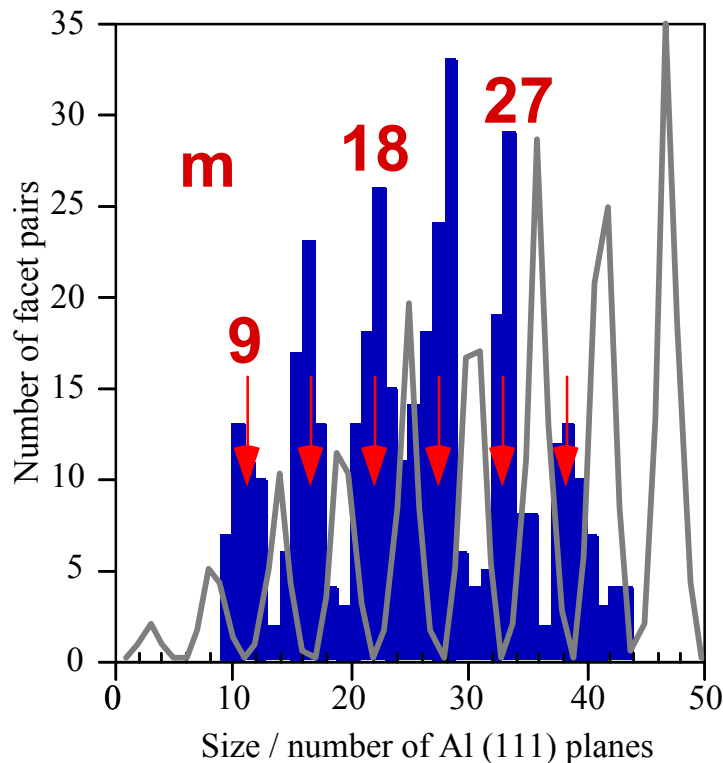
Building  
Blocks for Zero  
Strain  
Precipitates



16.75Å



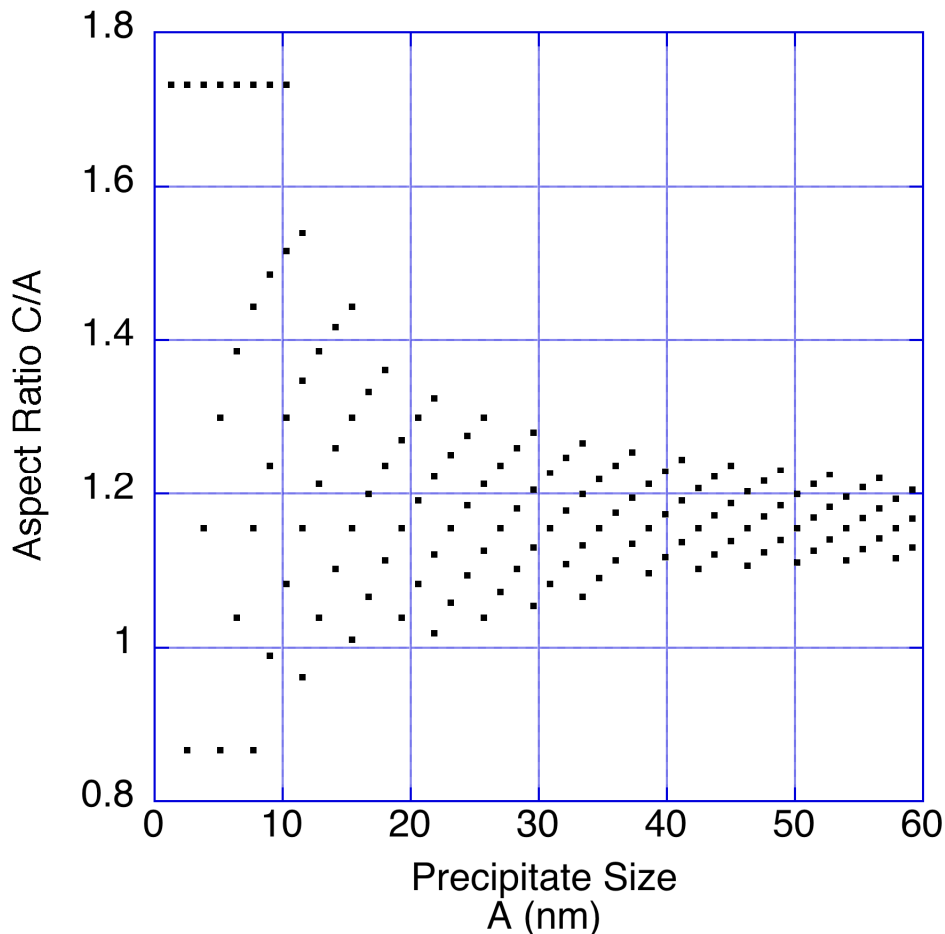
Building Blocks for Zero and  
Small Homogeneous Strain  
Precipitates



- Histogram shows experimentally observed distances between  $\{111\}$  planes.
- 16.75 Å edge lengths allow building shapes with peaks with  $m \approx 12, 21, 30, \dots$
- The “magic shapes” criterion for small homogeneous strain precipitates replace the criteria of “magic sizes”.

Dahmen, Xiao,  
Paciornik, Johnson and  
Johansen, PRL 78, 471  
(1997)

# Theoretical Prediction

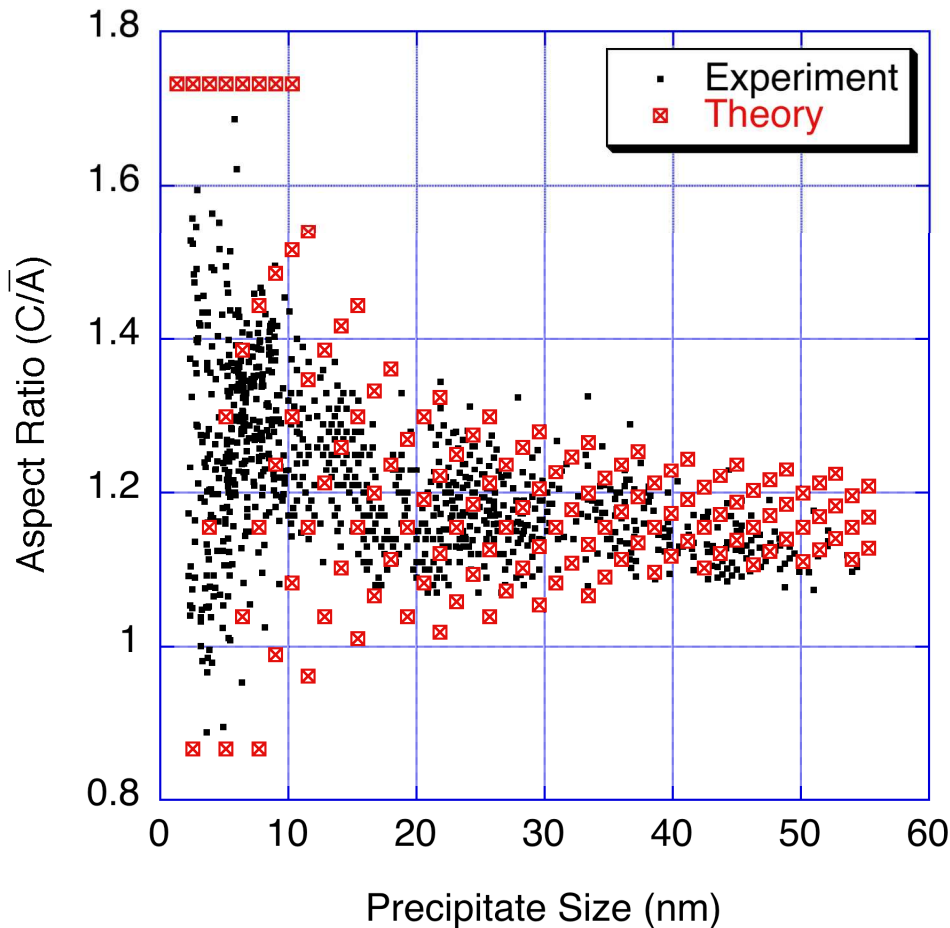


## • Method

- Generate set of “magic shapes” using square pyramids and tetrahedra with  $s=16.75\text{\AA}$ . Choose subset with  $O_h$  symmetry.
- Strain energy for these shapes is small, so we use approximation,  $E_{\text{precipitate}} = A_{111}\gamma_{111} + A_{100}\gamma_{100}$ .
- Calculate  $E_{\text{Wulff}}$ , the energy of precipitate with same volume, but with Wulff shape.
- Plot Aspect Ratio for  $O_h$  magic shapes with  $\Delta E = E_{\text{precipitate}} - E_{\text{Wulff}} < 100\text{eV}$ .
- This energy criteria is not rigorous, but appears to account for system which while approaching equilibrium is still far from equilibrium.



# “Magic-Shape” Theory Agrees With Experiment



## • Conclusions

- Size-dependent shape effects for Pb in Al are not due to edge energy.
- These effects are explained by combined effect of strain and interface energies.
- Calculating edge energies and/or facet areas requires a rigorous definition of edge length. We propose Gibbs equimolar surface as a solution to this problem.