

# Microstructure and Properties of PH13-8Mo Steel Fabricated by LENS®

Baolong Zheng<sup>1</sup>, John E. Smugeresky<sup>2</sup>, Yizhang Zhou<sup>1</sup>,  
and Enrique J. Lavernia<sup>1</sup>

<sup>1</sup>Dept. Chemical Engineering and Materials Science  
**University of California, Davis**  
Davis, CA 95616

<sup>2</sup>**Sandia National Laboratories**  
Livermore, CA 94551-0969



*Sandia National Laboratories is a multi-program laboratory operated and managed by Sandia Corporation, a wholly owned subsidiary of Lockheed Martin Corporation, for the U.S. Department of Energy's National Nuclear Security Administration under contract DE-AC04-94AL85000.*



# Outline

- Objective
- Background on LENS
- Microstructure and mechanical properties evaluation
- Thermal behavior during LENS® processing
- Summary



# Objective

- **Determine and understand microstructure-property relations, and thermal behavior of laser engineered net shaped (LENS<sup>®</sup>) PH13-8Mo stainless steel.**

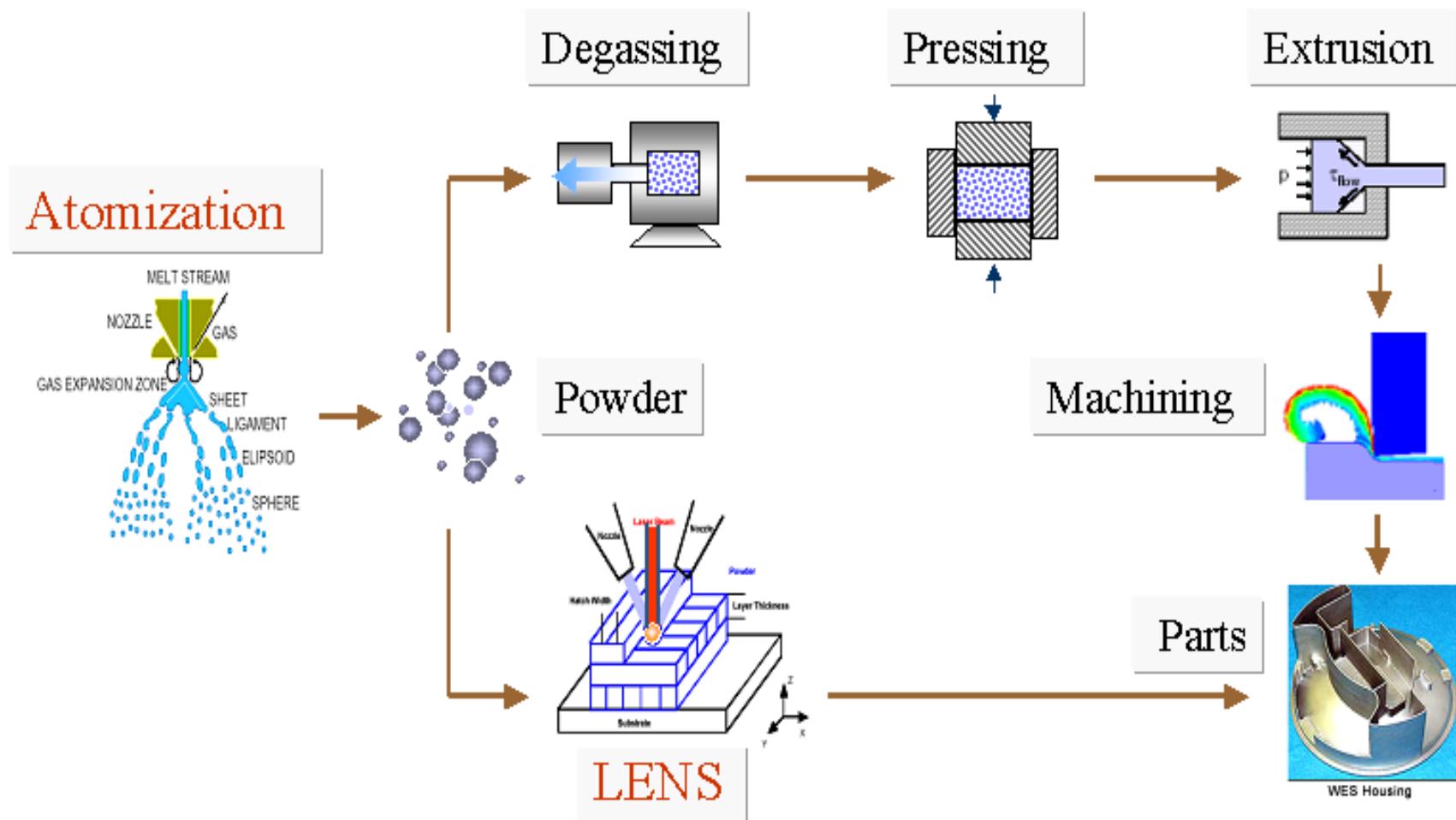


- PH13-8Mo Stainless Steel.
  - Is a low carbon (<0.03) precipitation hardenable, martensitic stainless steel.
  - Has the highest combination of corrosion resistance, strength and toughness in all the stainless steels
  - Is an Ideal choice for extreme environmental conditions
- Applications include forged airframe parts, fasteners, undercarriage, petrochemical, and safety and security components.

PH 13-8 Mo stainless is a registered trademark of Armco Inc.

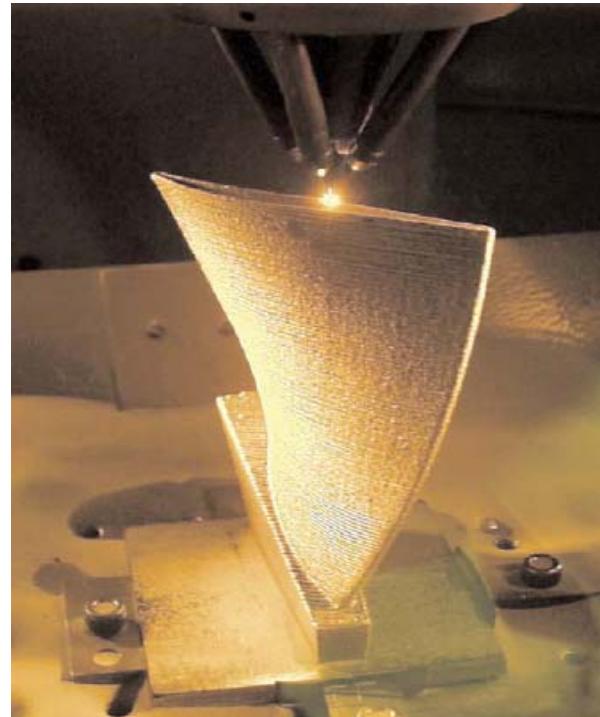
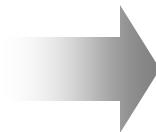
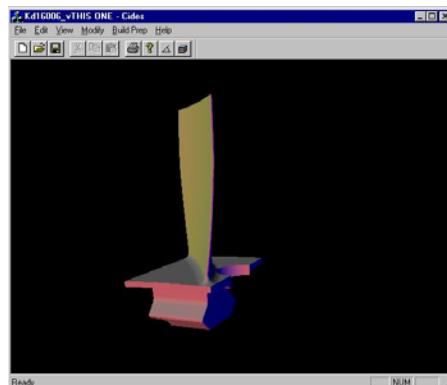
# Background on LENS

LENS® Is an alternative technique for consolidating metal powders to net-shape



# Laser Engineered Net-Shaping (LENS®) Details

- A novel rapid manufacturing process: Using a focused high-powered laser beam to melt the injected powder to net shape components.



- CAD solid model
- Electronically slice to horizontal layers
- Automatically controlled laser powder deposition
- Forms part line by line, layer by layer
- Without tooling

SNL, 2001

# Advantages & Applications

## Advantages:

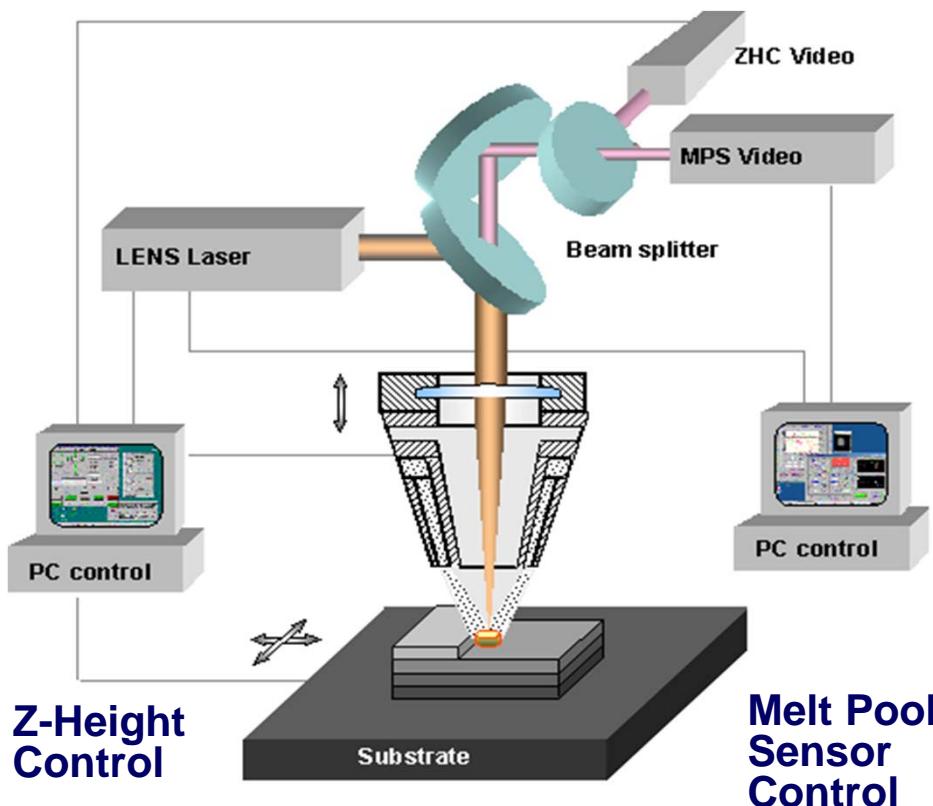
- Small heat affected zone with high cooling rate.
- Excellent material properties due to fine microstructure features.
- Easy composite and functionally graded material deposition with high accuracy.
- Fully dense net shape part deposition bypassing initial forming operation .
- Flexibility and accurate manufacturing.
- Minimal material waste.

## Applications:

- **Rapid prototypes**
  - Functional, smart
- **Modification & repair**
  - Failures in field , fabrication
- **Small lot production**
  - Spare parts, new / development parts
- **Improve properties of existing alloys and composites**
- **Develop new alloys and composites**

# Components of LENS® System used for This Study

- LENS® workstation 750 equipped with in-situ melt pool size sensor (MPS) control and Z-height sensor control (ZHC) close-loop subsystems.



- CW Nd:YAG laser (1064 nm) to create a melt pool
- 4-nozzle coaxial powder feed system
- 5-axis positioning control system assisted by CAD model
- Controlled environment glove box
- Energy density in the range of 30,000-100,000W/cm<sup>2</sup>

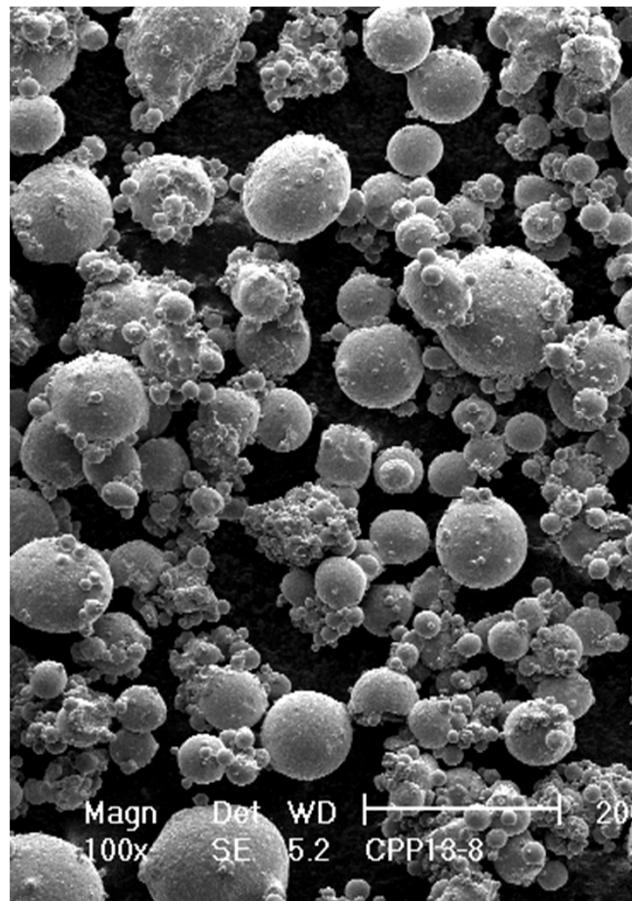
# Outline

- Objective
- Background on LENS
- **Microstructure and mechanical properties evaluation**
- Thermal behavior during LENS® processing
- Summary

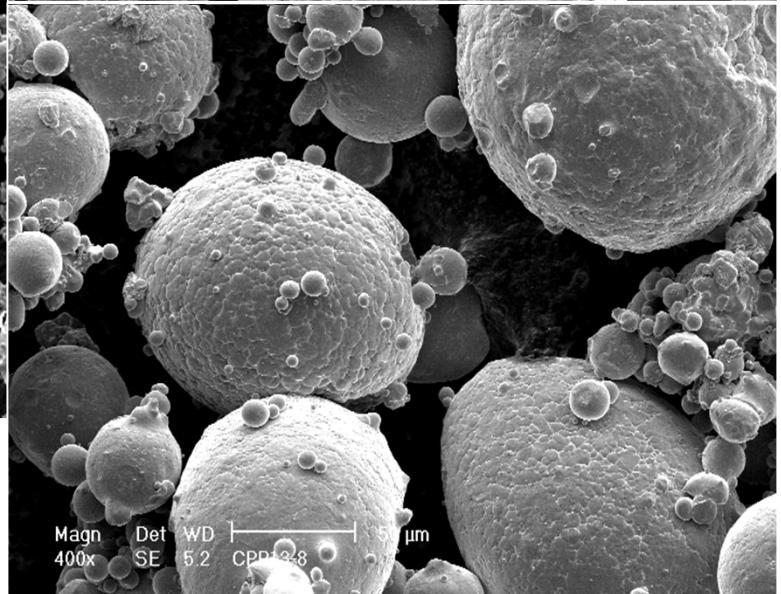
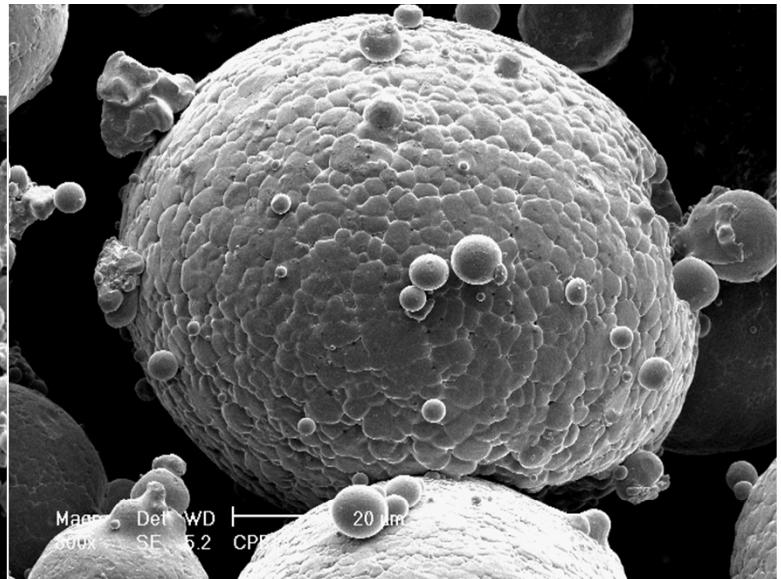
# PH13-8Mo Powder

## Chemical Composition

Sym.	wt% in Product
Fe	Balance
Cr	12.6
Ni	8.2
Mo	2.2
Al	1.1
C	0.026
Mn	0.01
Si	0.02
P	<0.005

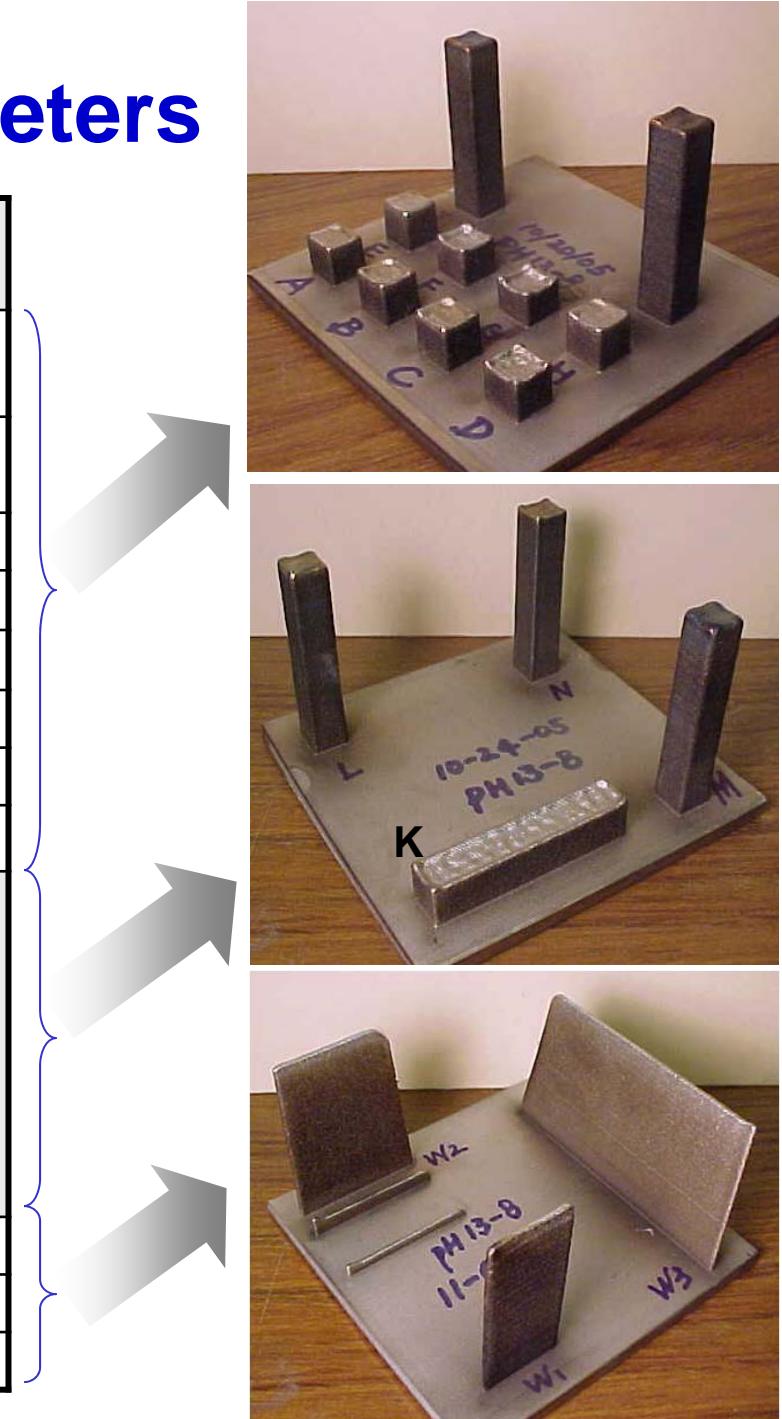


Gas atomized powder  
-100+325mesh (45-150μm)

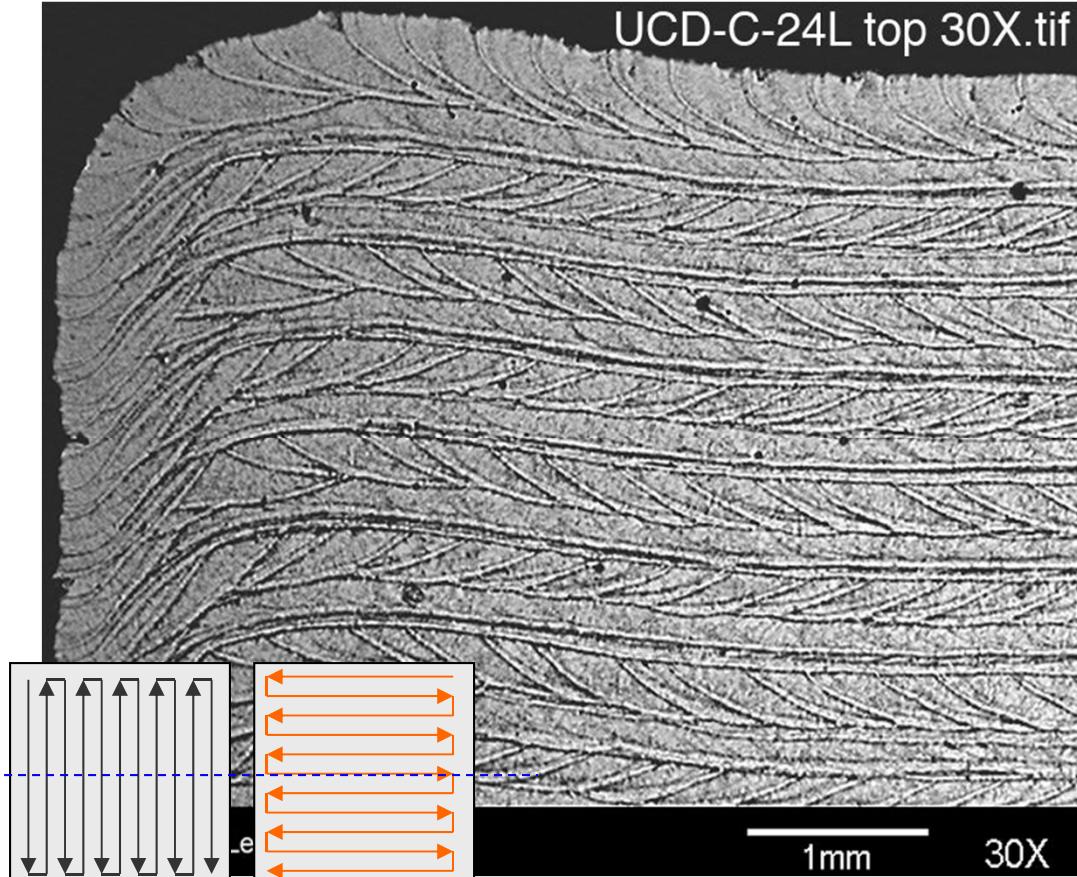
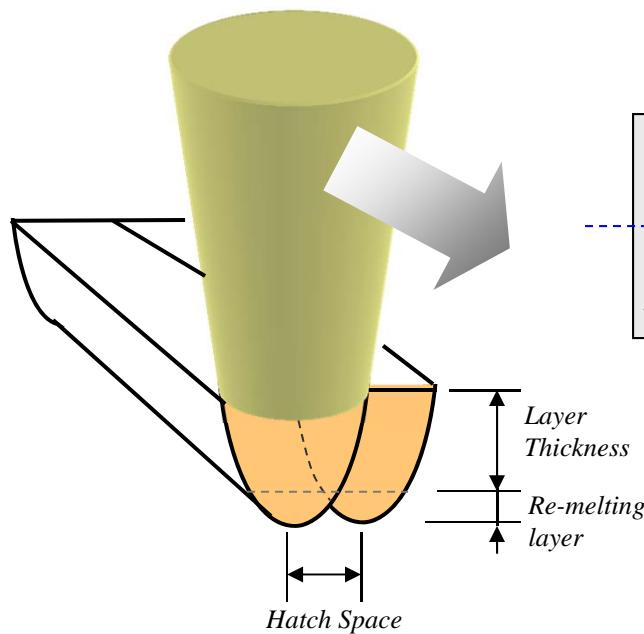
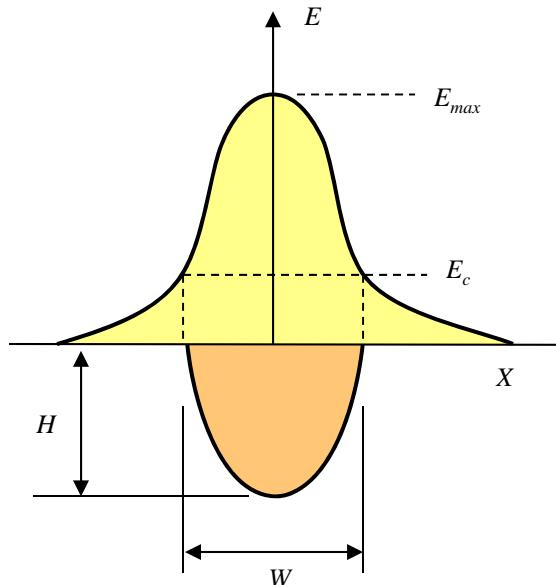


# Samples & Process Parameters

Shape	No.	Power (W)	Speed (mm/s)	Others
<b>Cubic 10x10 x10 mm</b>	A	355	14.8	<b>A/Deceleration = 847mm/s<sup>2</sup></b>
	B	355	14.8	<b>A/Deceleration = 212mm/s<sup>2</sup></b>
	C	355	14.8	<b>Shutter Delay 100 ms</b>
	D	270	12.7	
	E	440	16.9	
	F	355	14.8	<b>Hatch: 0.51 mm</b>
	G	355	14.8	<b>Lay thickness: 0.38 mm</b>
	H	355	14.8	Normal condition *
<b>Cubic 10x10 x51 mm</b>	I	355	14.8	
	J	355	<b>16.9</b>	
	K**	355	14.8	
	L	355	<b>12.7</b>	
	M	440	16.9	
	N	270	12.7	
<b>Thin wall</b>	W1	355	14.8	<b>Contour 22x51x0.25 mm</b>
	W2	355	14.8	<b>Single 38mm x 51mm</b>
	W3	355	14.8	<b>Single 83mm x 51mm</b>

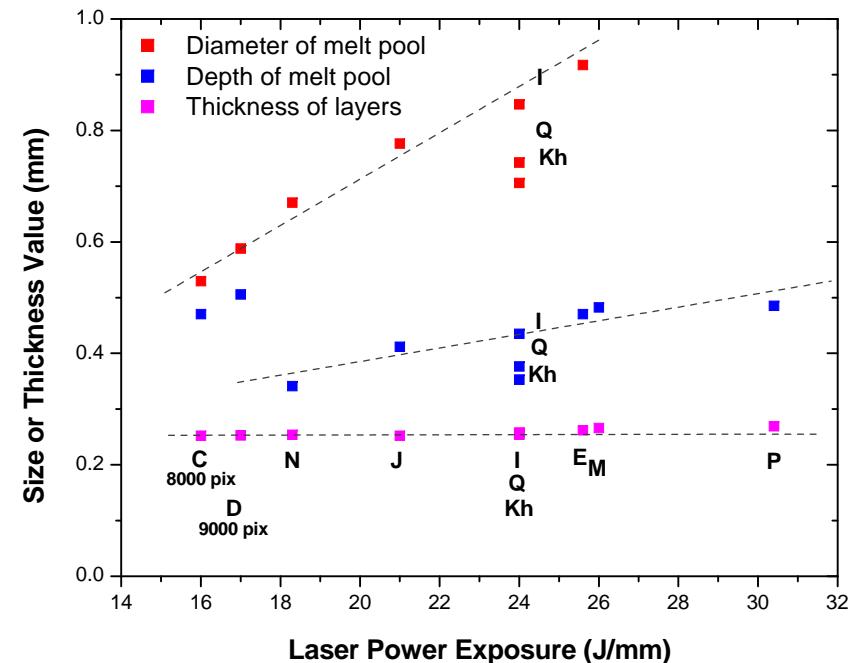
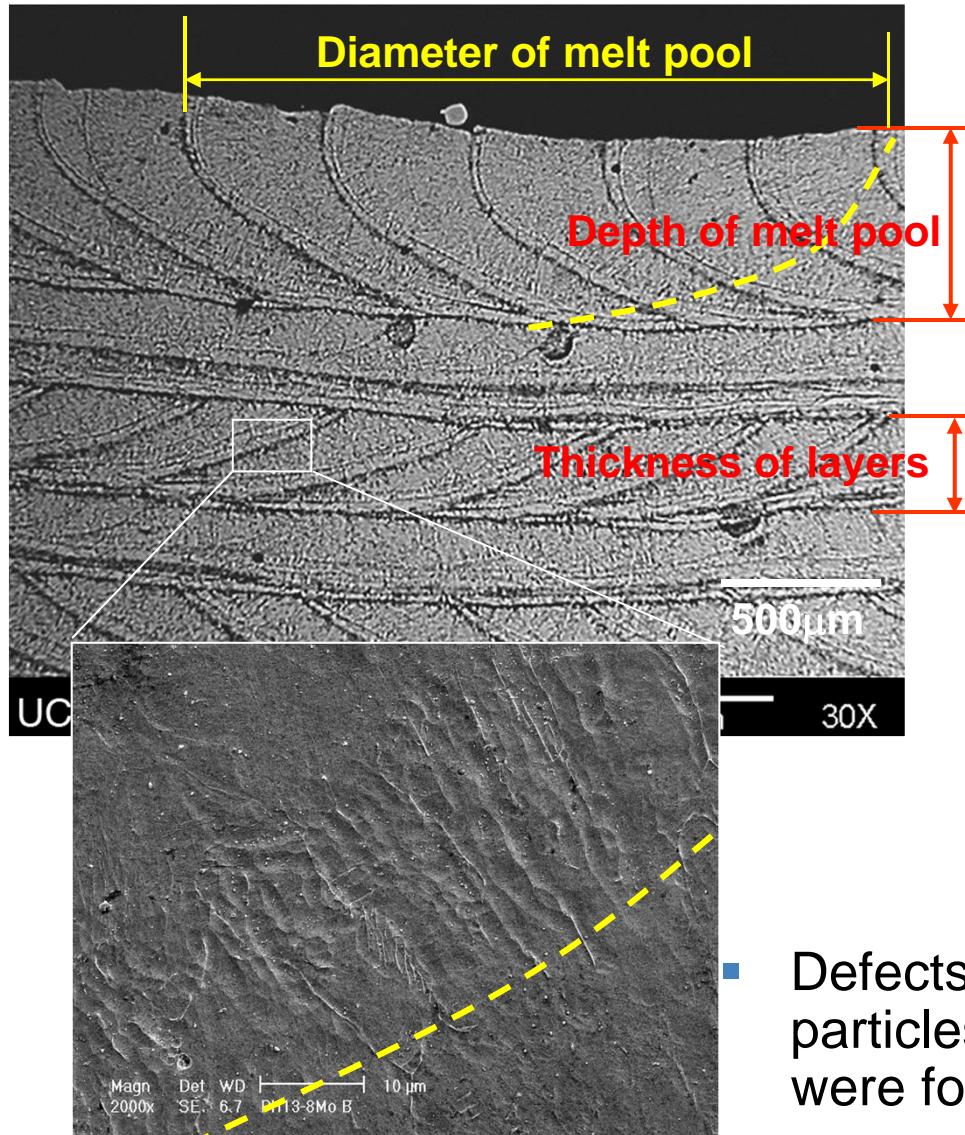


# Laser Beam and Melt Pool Track



- Melt Pool Track of overlap between adjacent lines and re-melting previous deposited layer

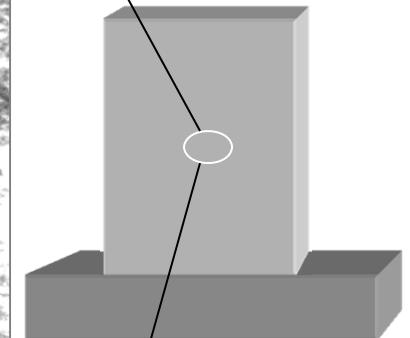
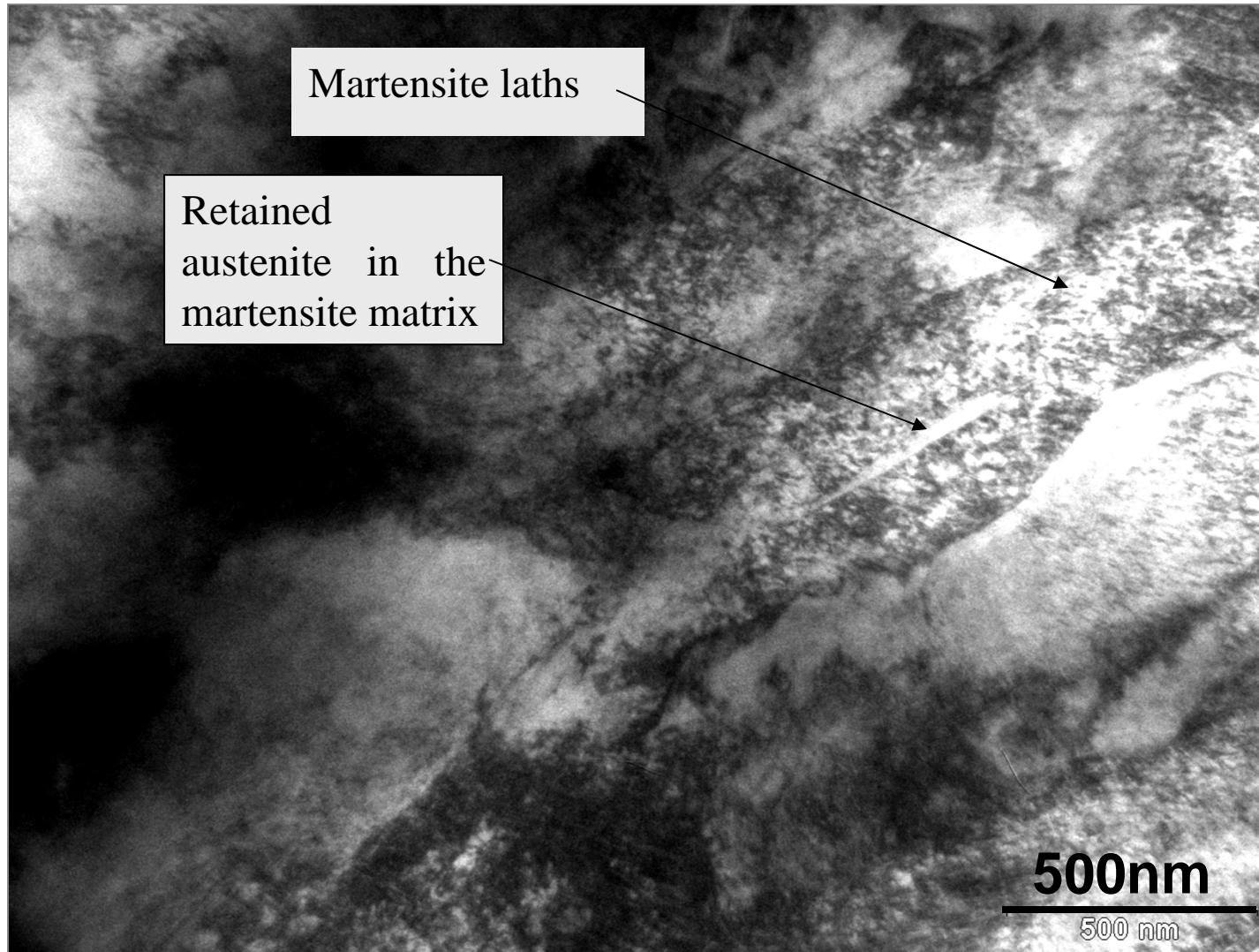
# Layer Features and Microstructure



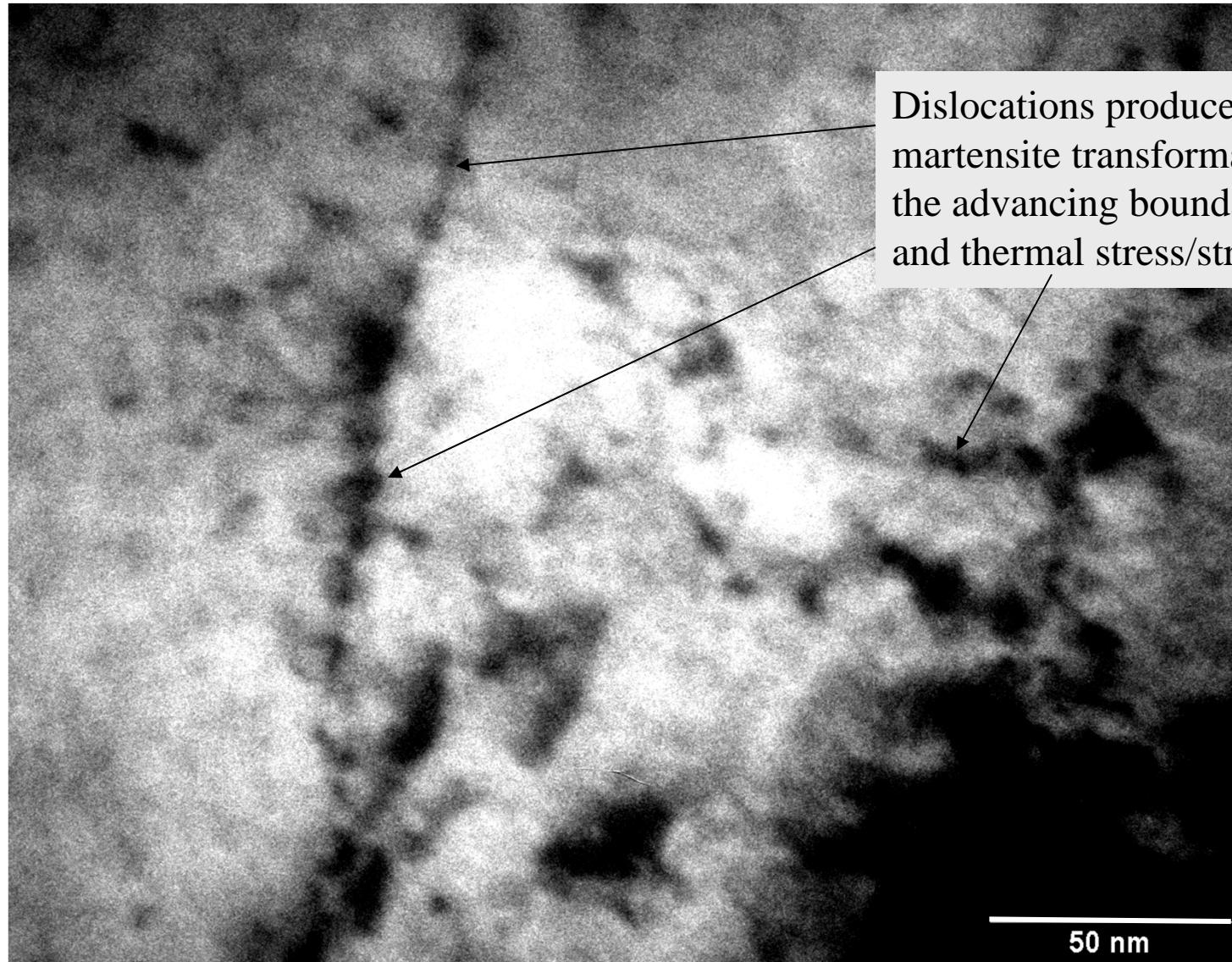
Dependence of melt pool diameter and depth on laser power exposure (power/speed).

- Defects of porosity and un-melted particles of around  $100\mu\text{m}$  in size were found

# Martensite Phase

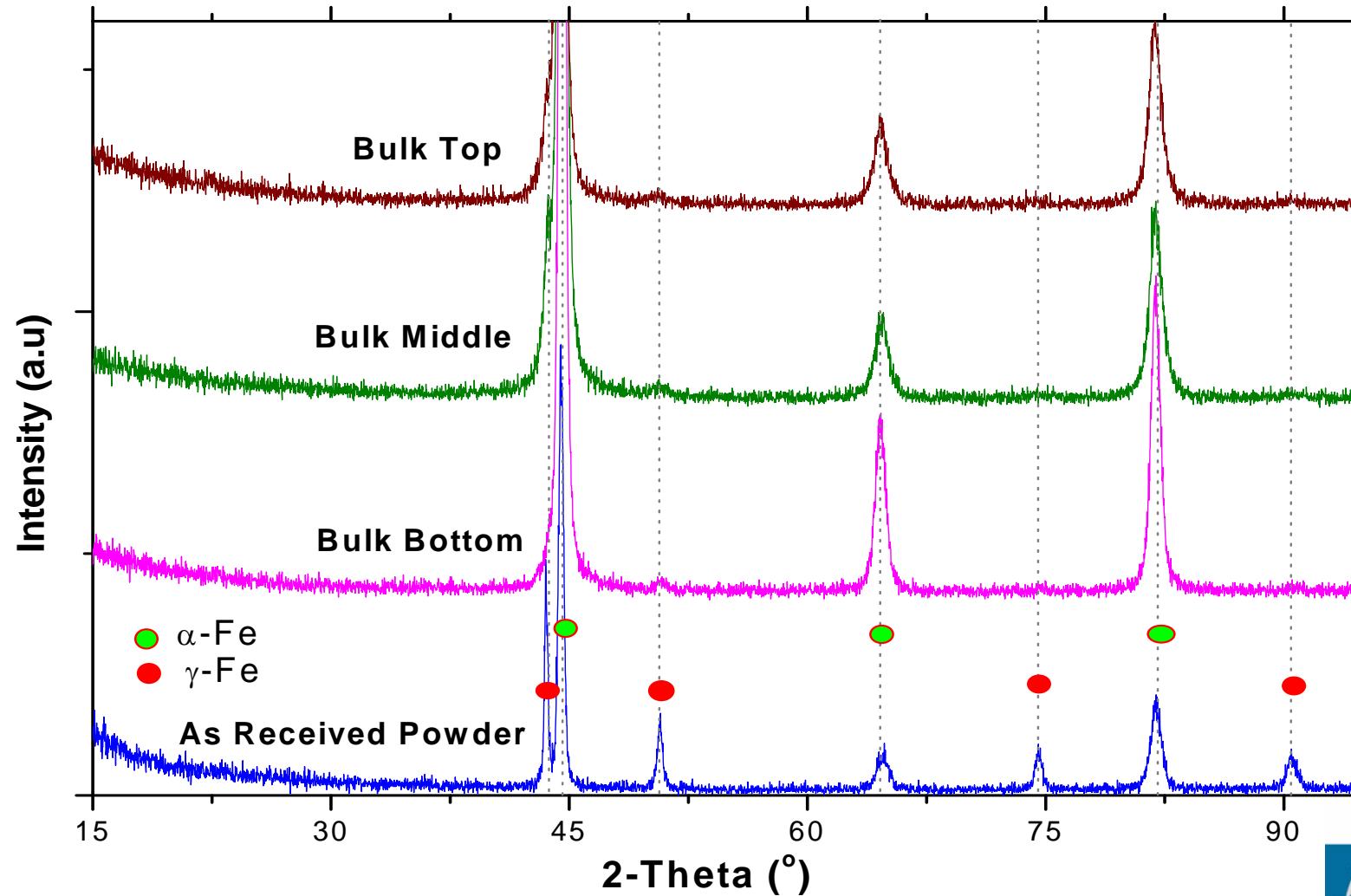


# Dislocations



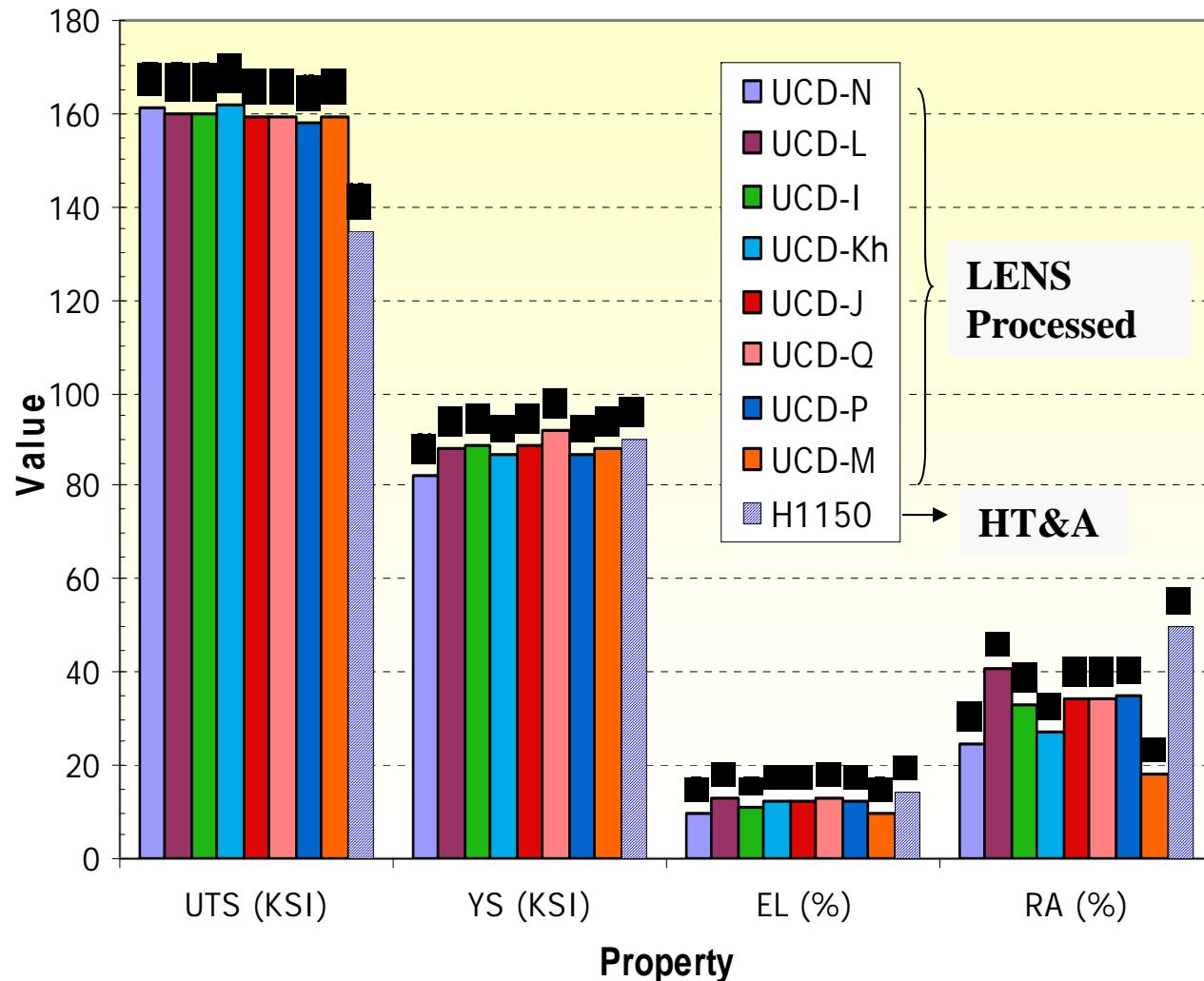
# XRD Diffraction

of LENS® deposited PH13-8Mo components and the gas atomized powder.



# Mechanical Properties of Cube Samples

Summary of LENS processed PH13-8Mo samples



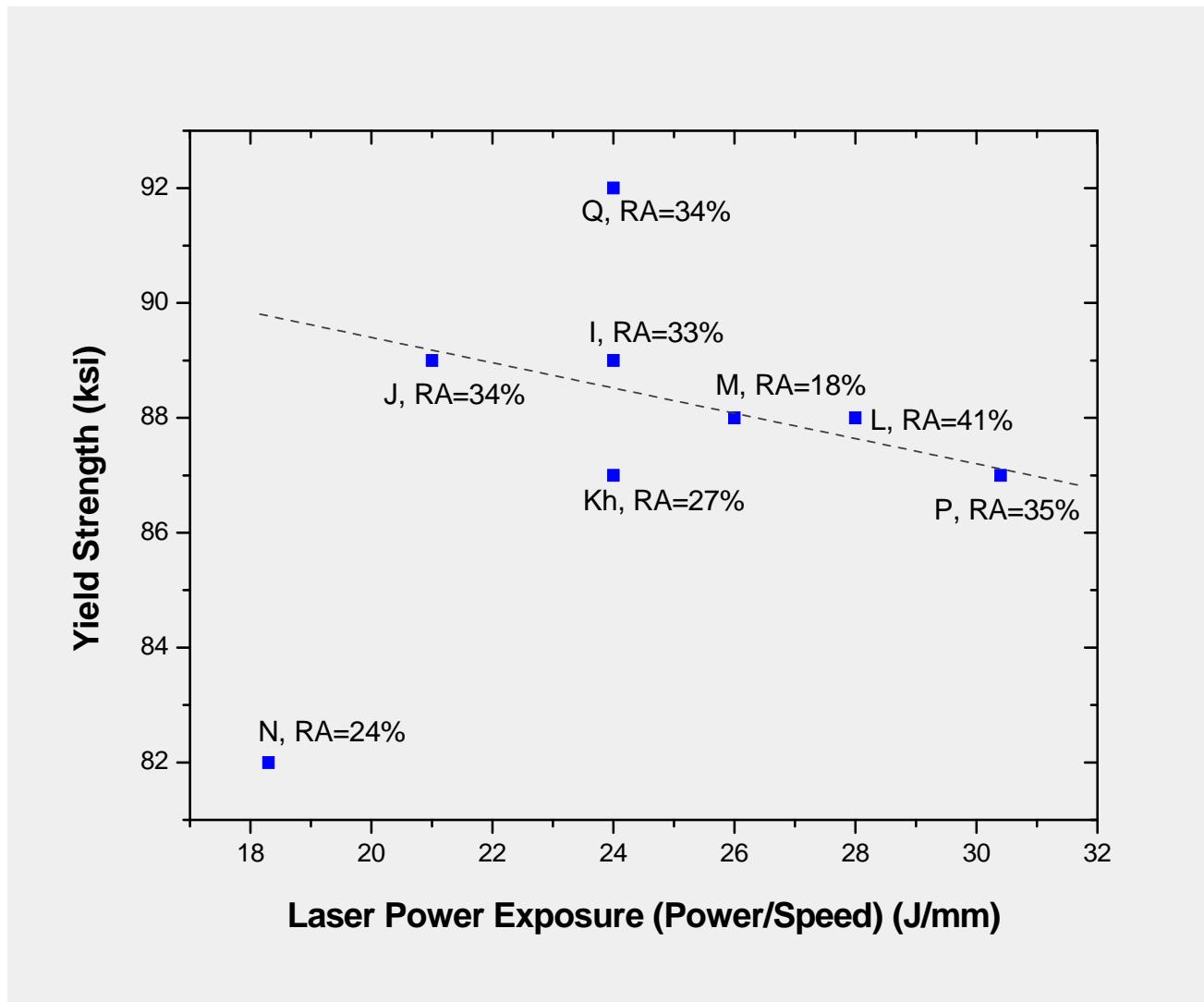
Properties	LENS processed	Conventional HT&A
UTS (ksi)	158–162	120–135
YS (ksi)	82–92	80–90
EL (%)	10–13	14–16
RA (%)	24–41	50–55

H1150:

Homogenization:  
1038° C (1900° F),  
1.5hr;

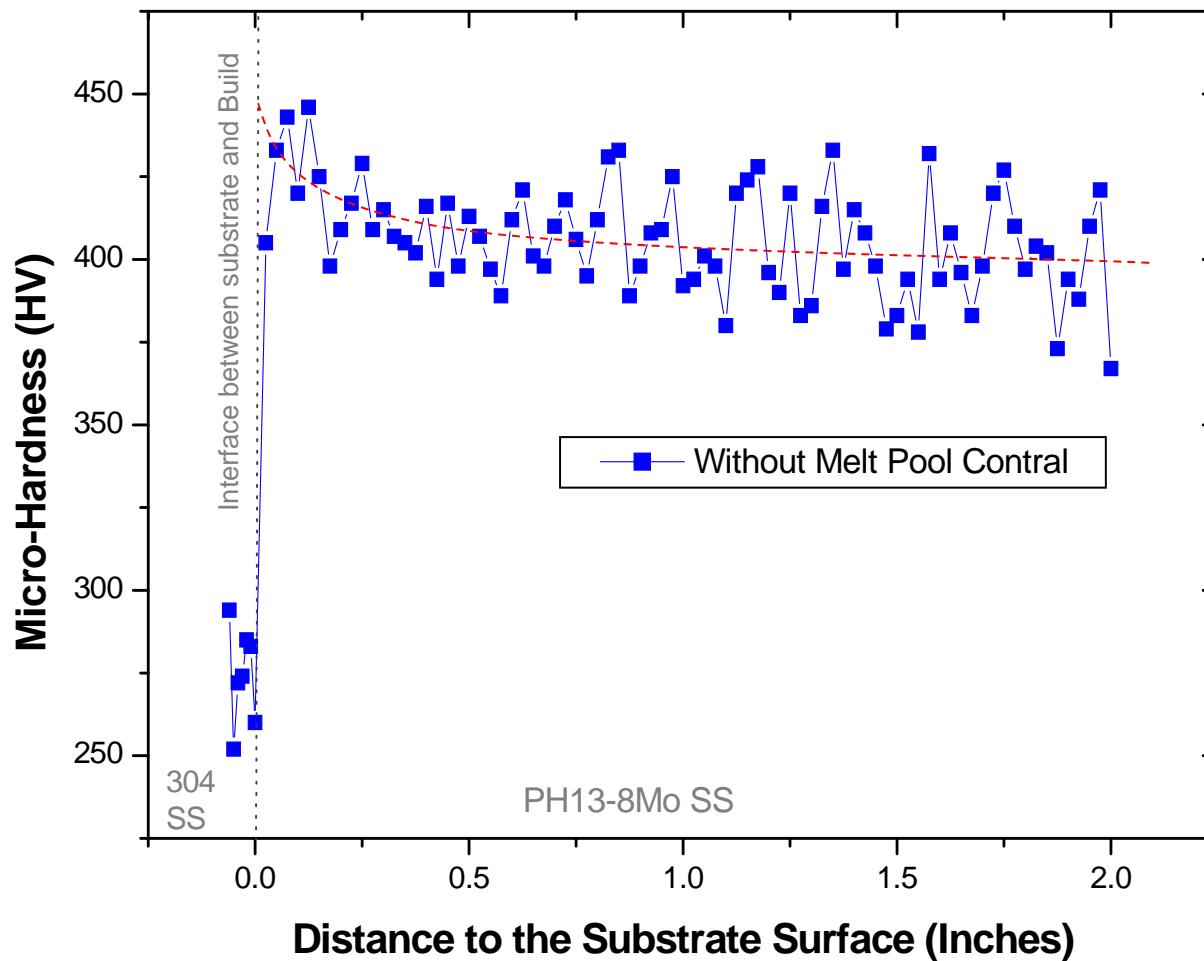
Aging: 621° C  
(1150° F), 4hr

# Effects of Process Parameters on YS



Yield strength vs.  
laser power  
exposure  
(power/speed).

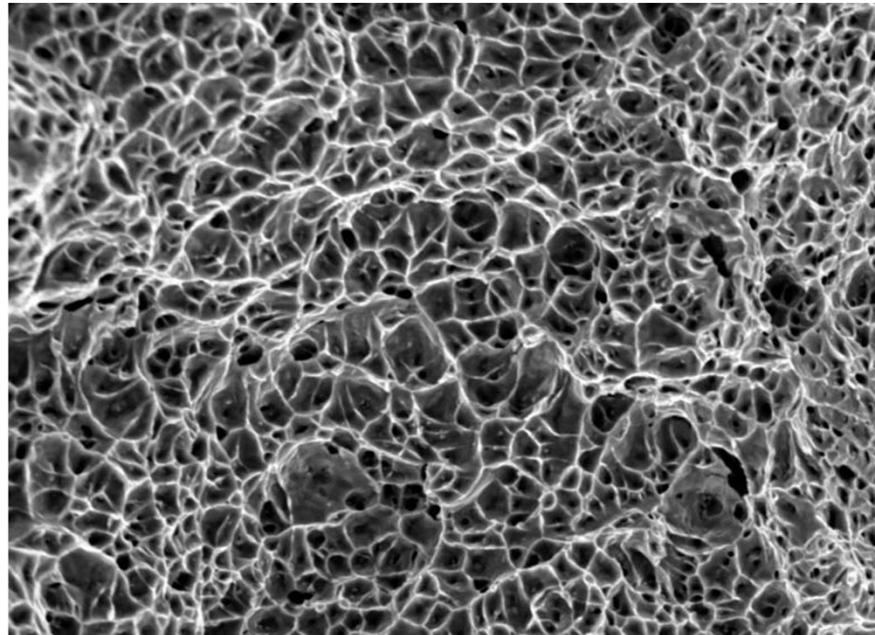
# Micro-Hardness



Tensile strength  $\approx$  0.33 x hardness,  
the mean tensile strength should be 200ksi (1373MPa).

LOP = 355 W  
LTS = 14.8 mm/s  
PFR = 10 g/min

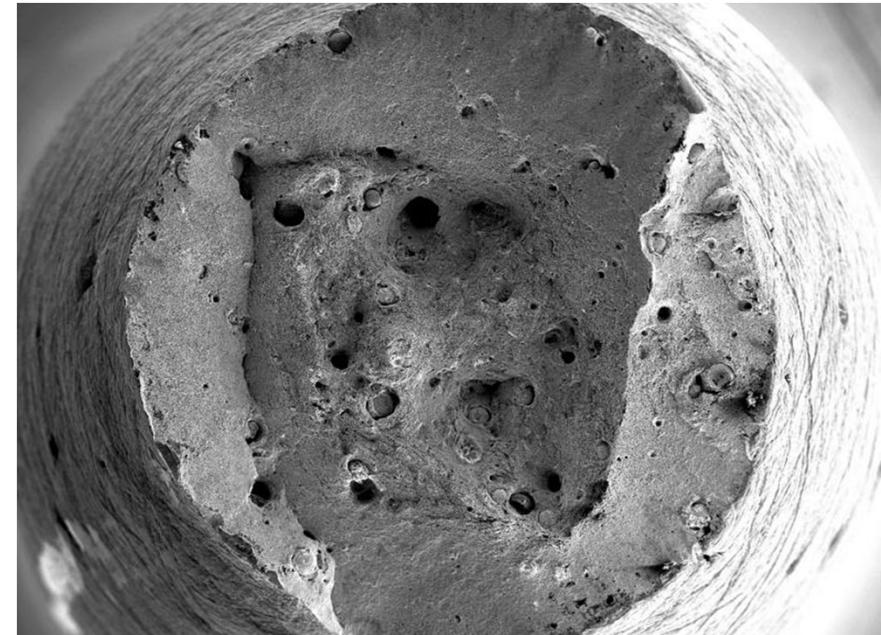
# Defects Initiated Ductile Fracture



M2000x.TIF

10 $\mu$ m 2000X

- Micro-dimple morphology
- Ductile fracture
- Crack initiation from particles at the bottom of dimples



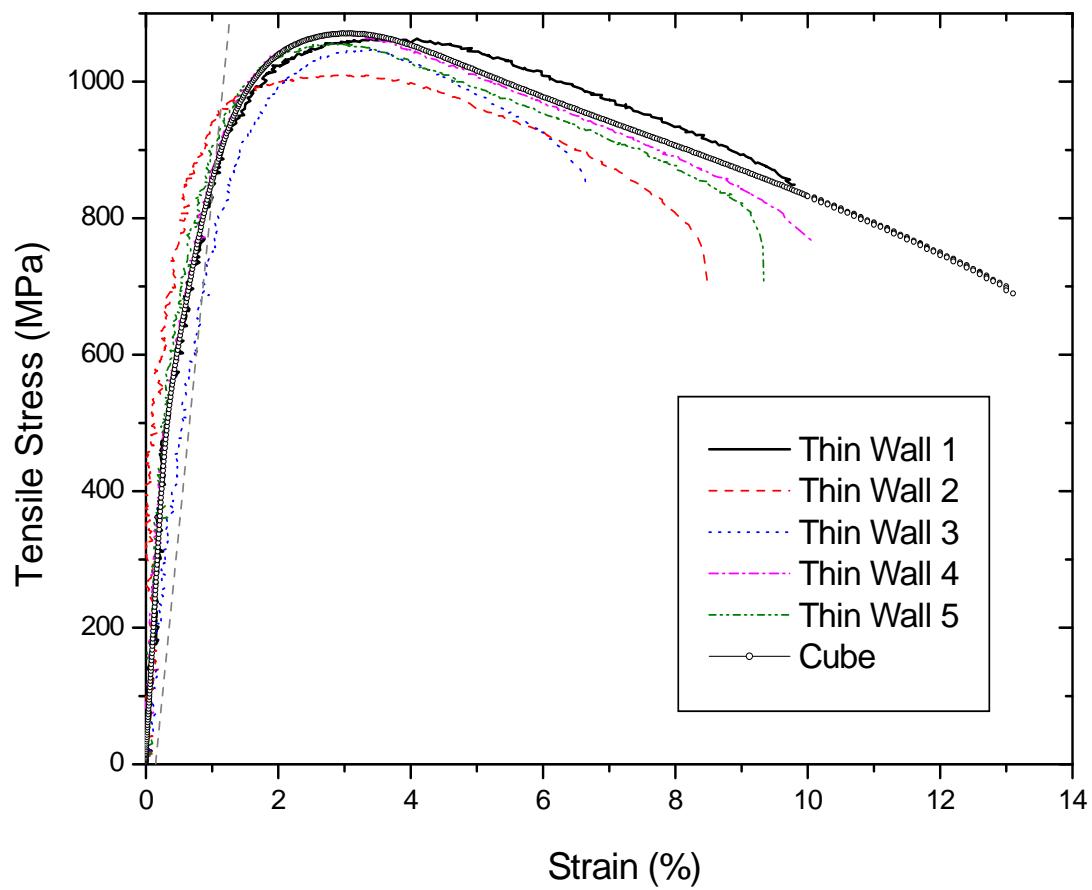
Q40x.TIF

600 $\mu$ m 40X

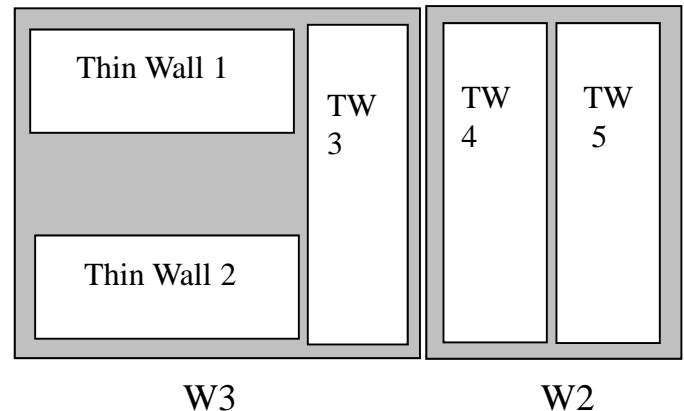
- Porosity and unmelted particles in the material, and observed on the fracture surface, affect strength

Premature fracture due to defects is the reason for low ductility

# Mechanical Properties of Thin Wall Samples



Stress-strain curves for LENS® deposited PH13-8Mo samples



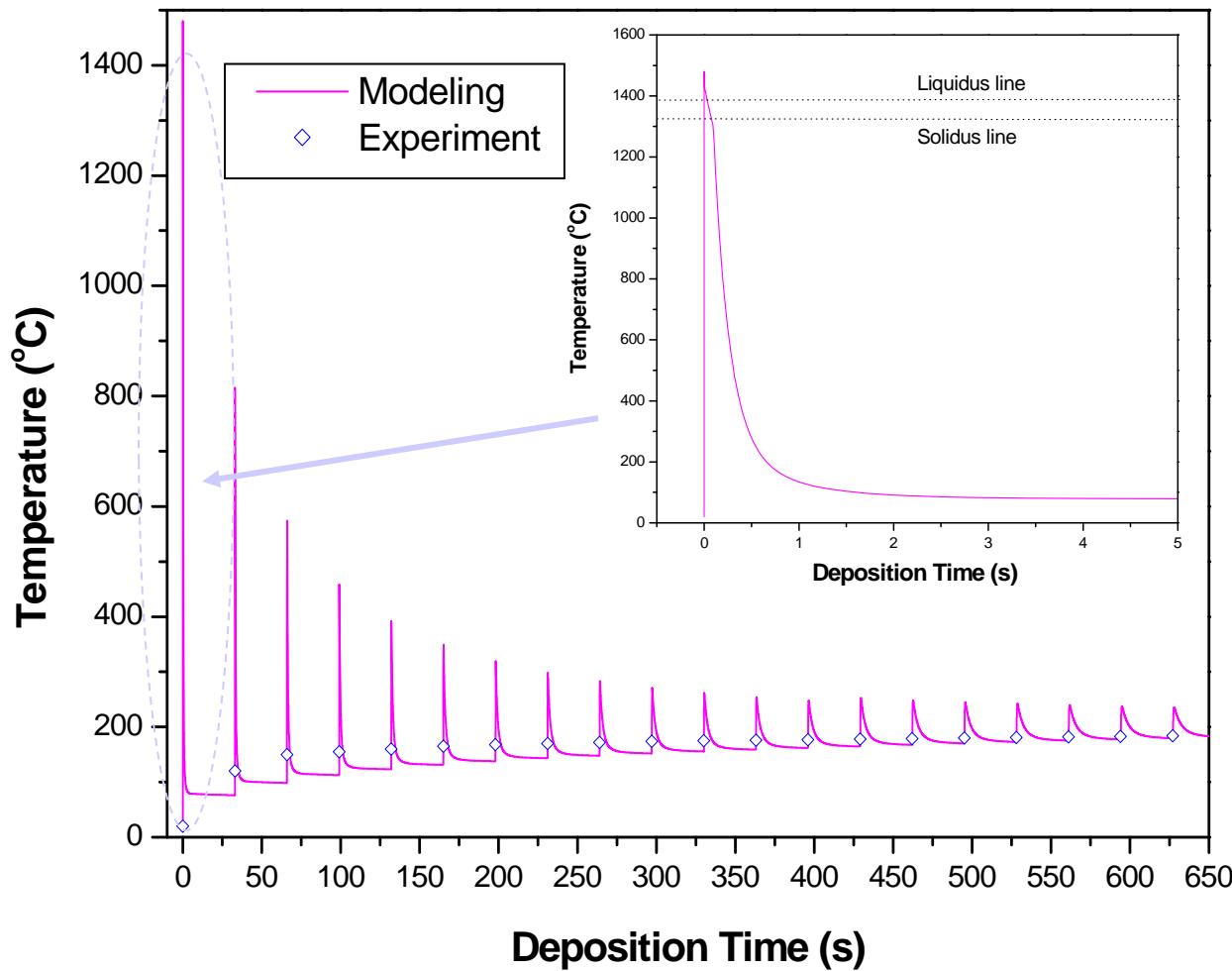
No.	YS (ksi)	UTS (ksi)	EL (%)
1	84	154	9.8
2	80	146	8.5
3	78	152	6.7
4	85	155	10.1
5	82	153	9.4
C	87	157	13.2

# Outline

- Objective
- Background on LENS
- Microstructure and mechanical properties evaluation
- **Thermal behavior during LENS® processing**
- Summary



# Thermal History during LENS® Processing



Thermal behavior associated with the LENS process involves numerous reheating cycles. As a result, the temperature history of the deposited materials is a dampen pulse wave.

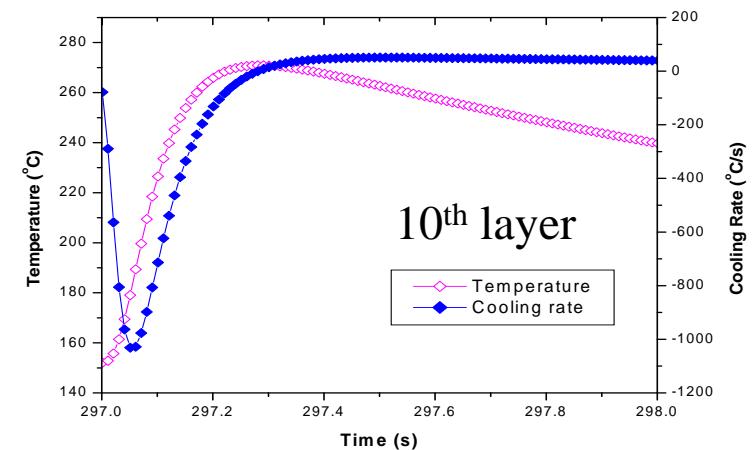
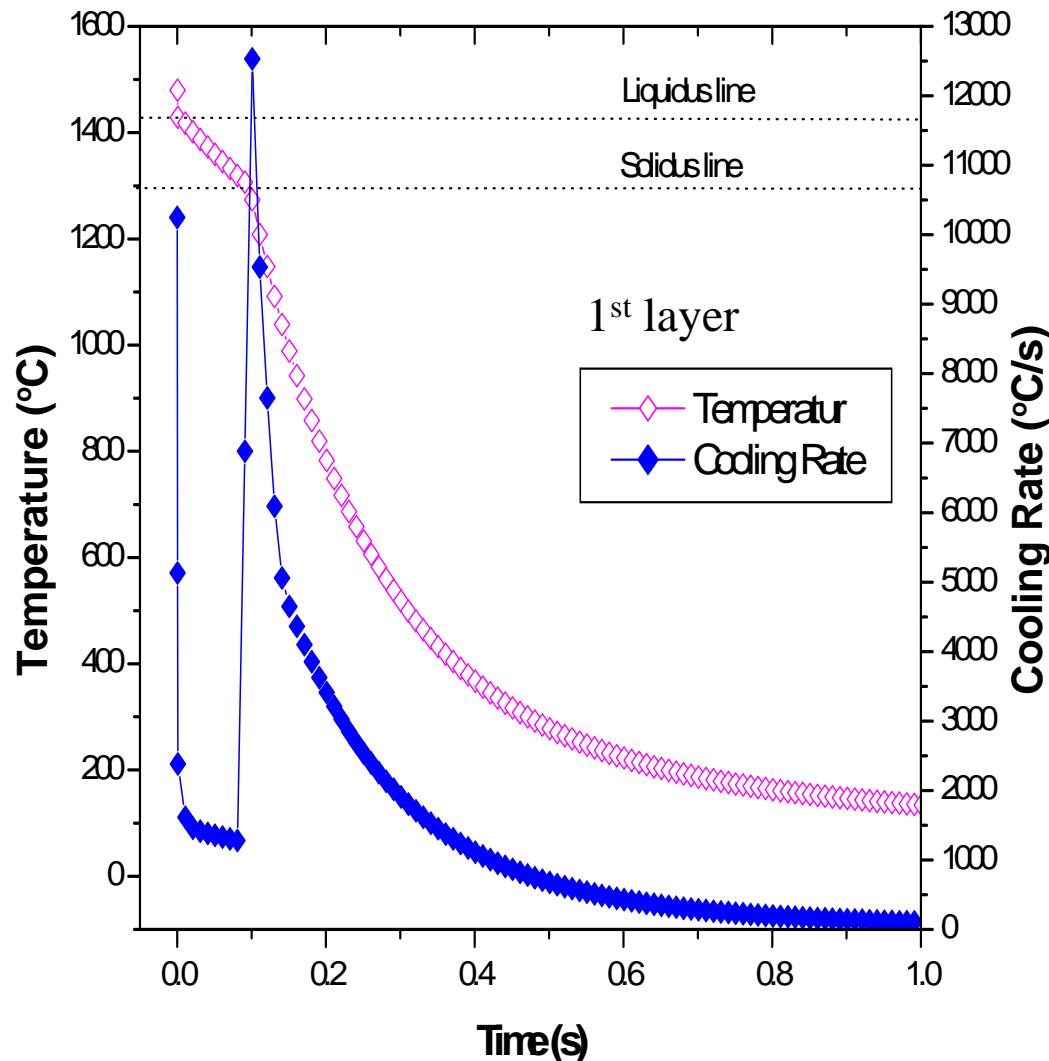
$$T_{mp} = 1480^\circ\text{C}$$

$$T_s = 20^\circ\text{C}$$

$$\Delta t = 33\text{s}$$

# Variation of Temp. and Cooling Rate

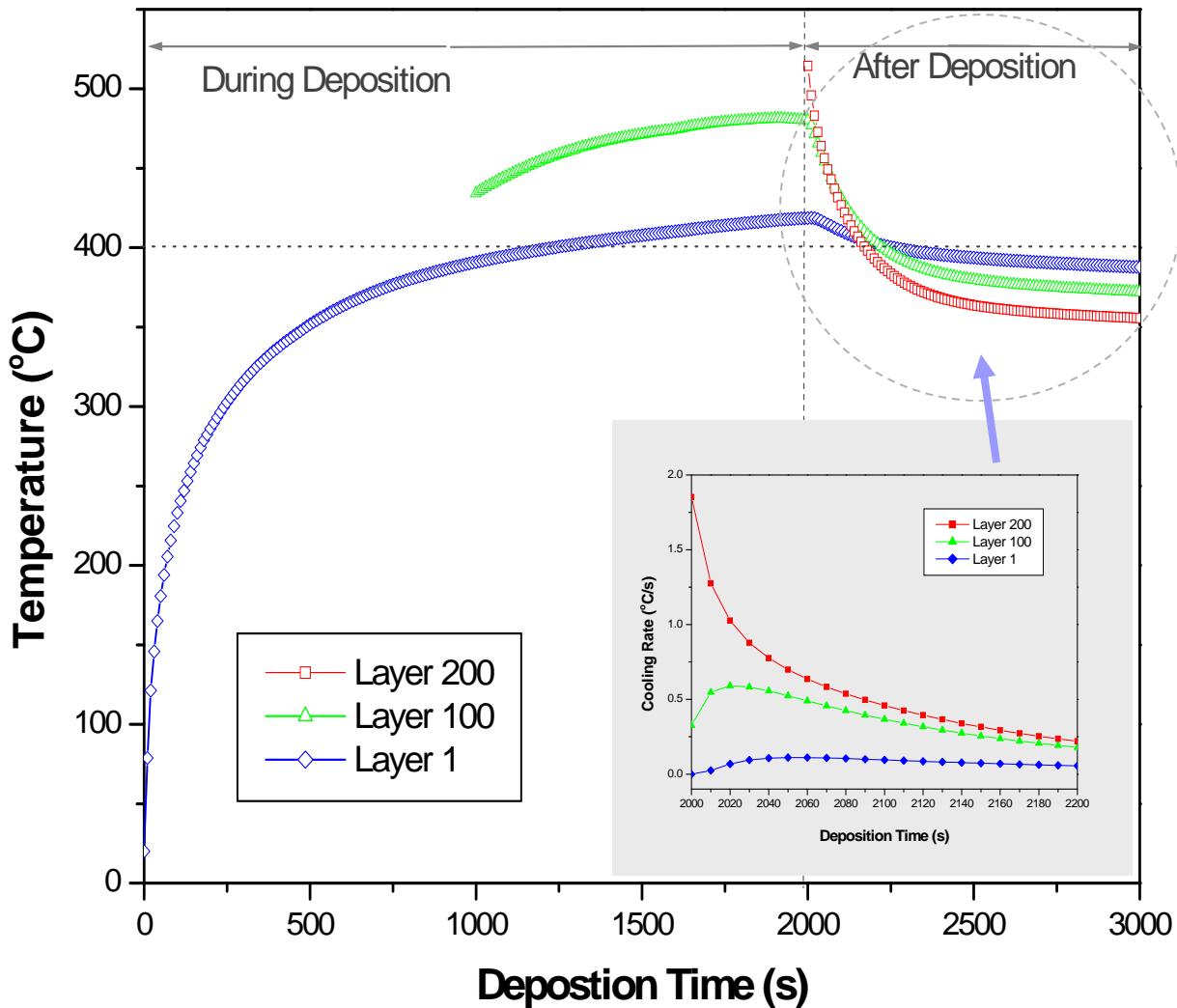
## during early stage of 1<sup>st</sup> and 10<sup>th</sup> layer deposition



In the initial stages of deposition, the deposited materials experience a significant rapid cooling effect, and can attain a very high cooling rate,  $\sim 10^4$ K/s.

In the later deposition, the rapid cooling effect decreases and even disappears.

# During and After Deposition



Variation of temperature at different position during and after deposition

$$\begin{aligned}T_{mp} &= 1480^{\circ}\text{C} \\T_s &= 20^{\circ}\text{C} \\\Delta t &= 10 \text{ s}\end{aligned}$$

Reheating cycles have tempering and aging effects; When temperature  $> 400^{\circ}\text{C}$ , NiAl particles will precipitate coherently with the matrix, leading to strengthening.

# Summary

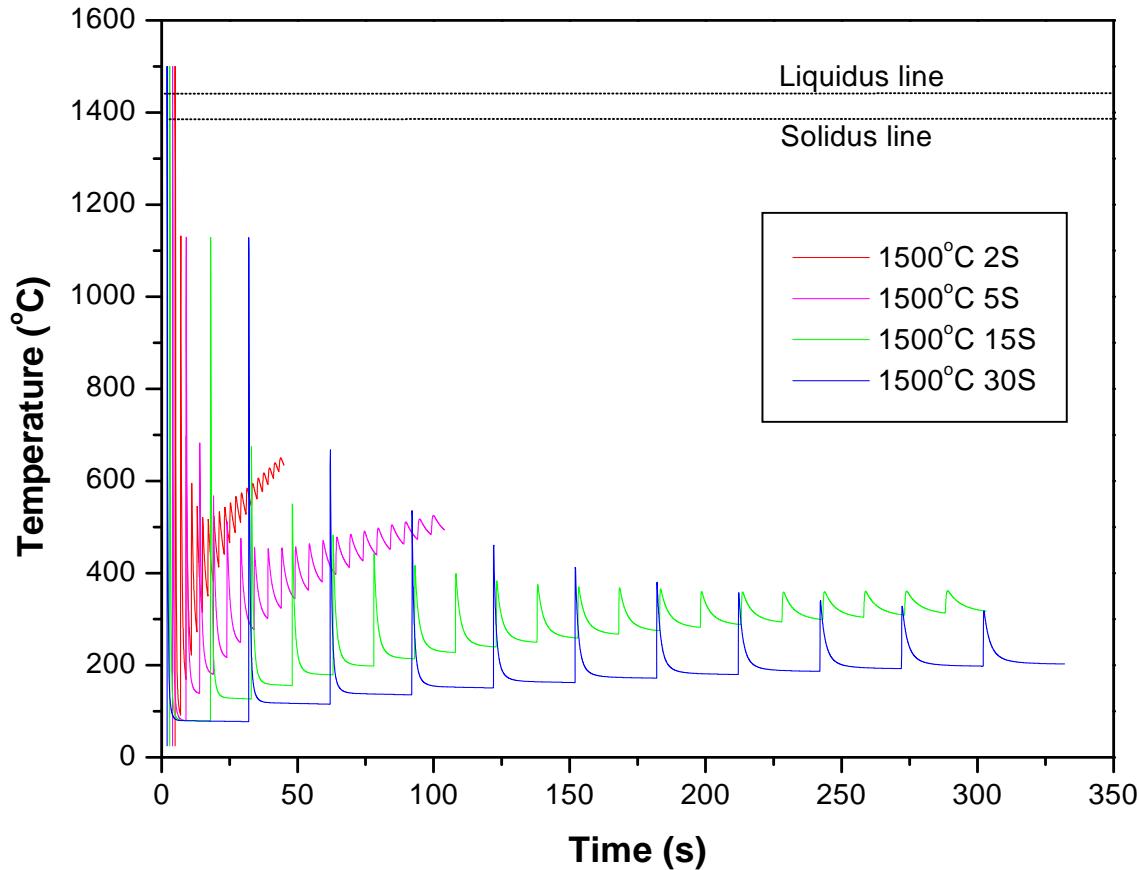
- LENS® process evaluated for making net shaped components from gas atomized PH13-8Mo powder;
- The mechanical strengths of the LENS® deposited PH13-8Mo material were equivalent to the heat treated and aged wrought materials;
- Microstructure contains  $0.2\mu\text{m}$  wide martensite laths nearly parallel with retained austenite. High density of dislocation and fine lath size is responsible for the high hardness
- Porosity and un-melted particles in the laser deposited materials caused reduced ductility, despite presence of a ductile dimple fracture mode;
- Thermal behavior associated with the LENS process involved numerous reheating cycles, which may have temper and aging effects, and may promote NiAl particle precipitation.

# Acknowledgments

- Work at UC Davis is partially supported by the U.S. National Science Foundation under grant DMI-0423695.
- Work at Sandia National Laboratories is supported by the U. S. Department of Energy under contract DE-AC04-94AL85000.
- Metallography and fractography were performed by A. Gardea and J. Chames respectively; Tensile tests were performed by M. Tootle and Dr. Bing Q. Han.

## Thanks for Your Attention

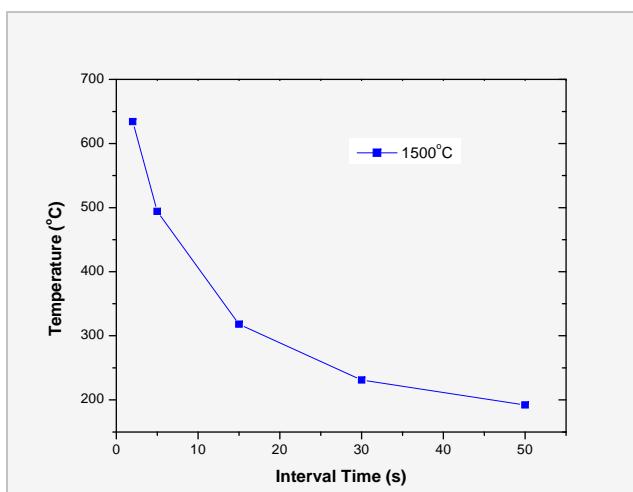
# Effects of Interval Time



Variation of thermal behavior with change of interval time

- Travel speed
- Part dimension
- Laser stay time between layer

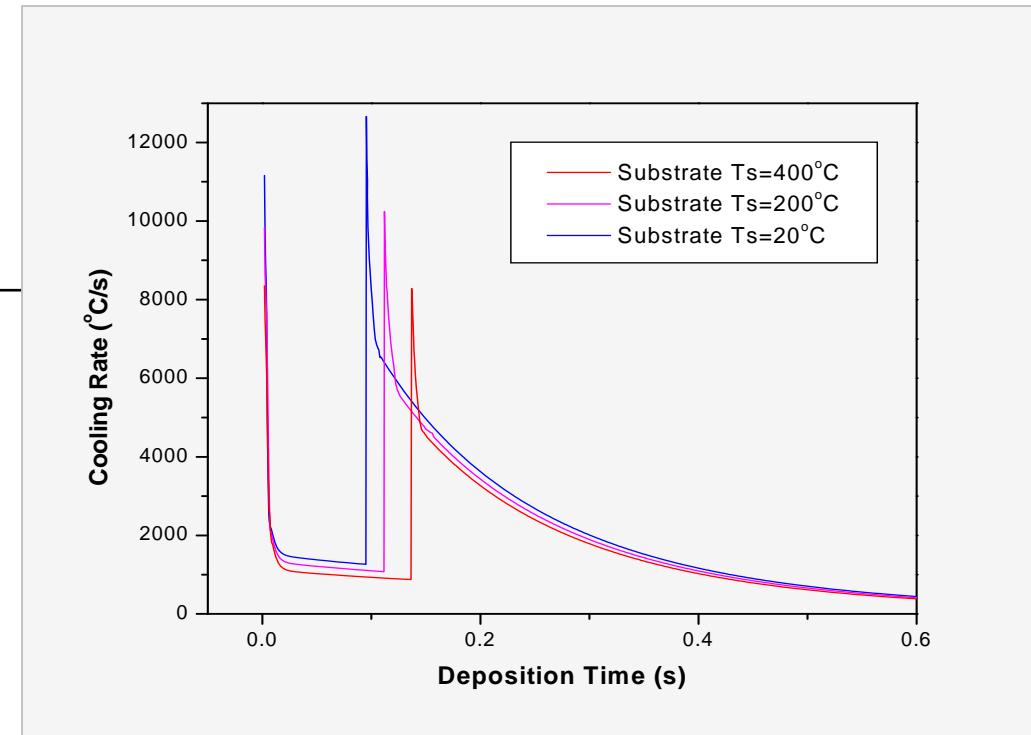
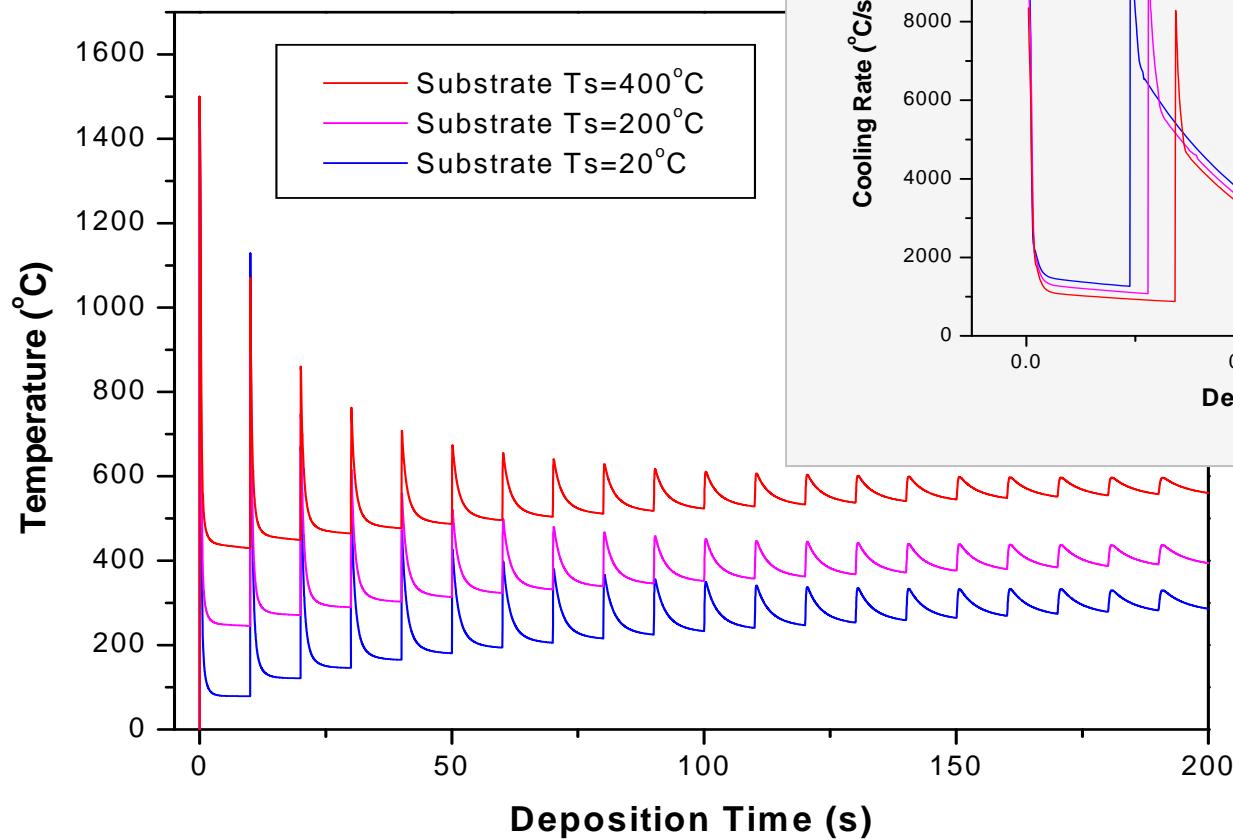
- Interval time



7

# Effects of Initial Temp. of Substrate

Variation of thermal behavior with initial temperature of the substrate



$$T_{mp} = 1500^\circ\text{C}$$
$$\Delta t = 10 \text{ s}$$