

Measuring Residual Stress in Glasses and Ceramics using Instrumented Indentation

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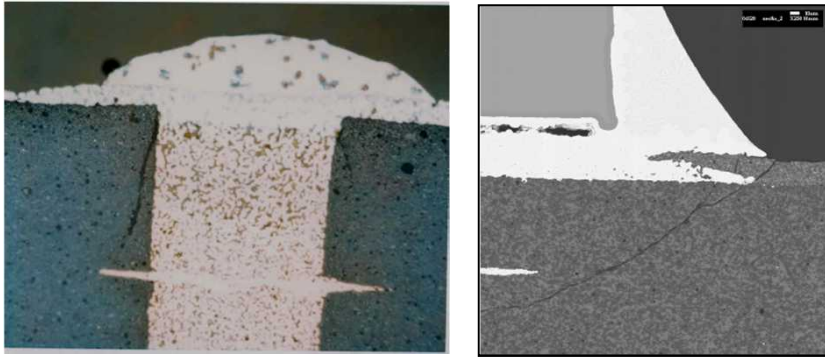
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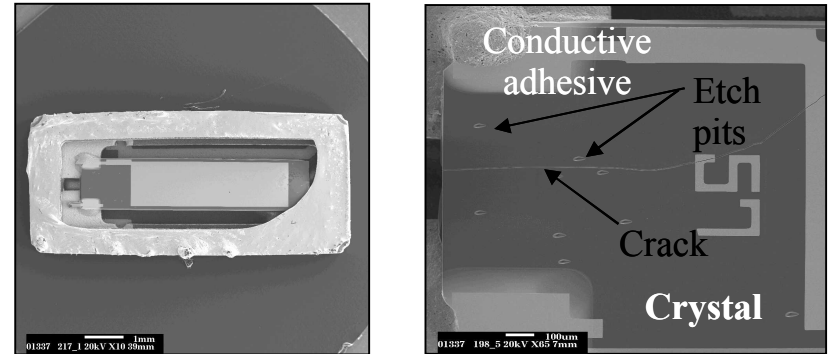
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Problem: Processing and thermally induced residual stresses lead to failure of ceramic-based microsystems components

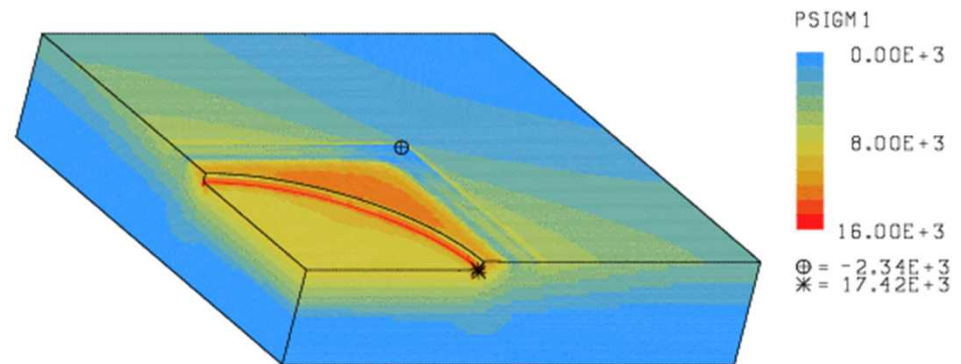
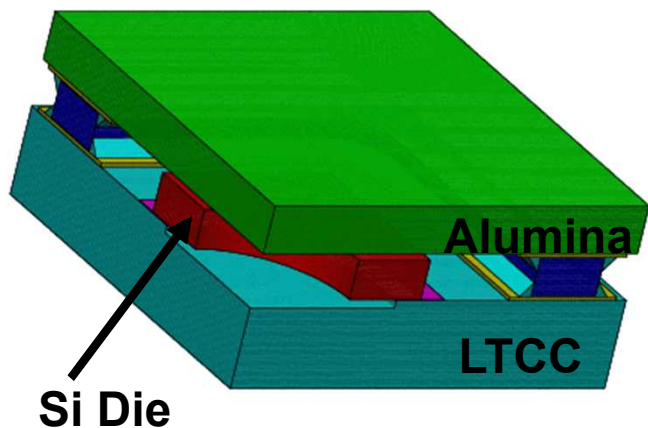
Examples of cracks in LTCC ceramic substrates



Quartz AT Strip SAW Device



FE simulation of Die Attached Microsystems assembly

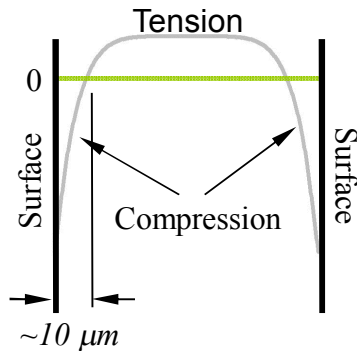


Can nanoindentation be used as a method to measure residual stress in brittle materials?

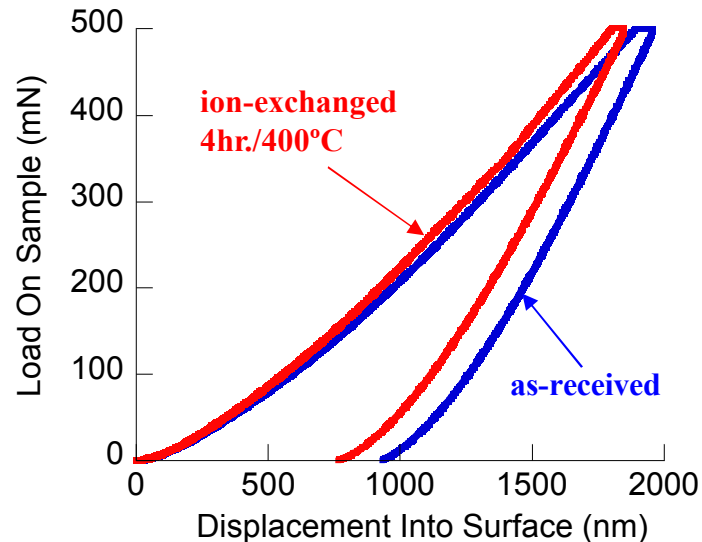
An initial observation:

- A very noticeable and repeatable difference in measured indentation response between unstressed and stressed glass.
- $E/\sigma_y \approx 25$ in glass, $E/\sigma_y \approx 150$ or greater in most metals.

Stress Profile in Ion-Exchanged Glass



PPG Glass – 500 mN load spherical tip- 10 μm nominal radius



A series of finite element simulations were performed to investigate the role residual stress on the indentation response

- 8 simulations- displacement control to 1 μm depth

material properties	tip geometry	residual stress
$E=72\text{ GPa}$ $\sigma_y=3\text{ GPa}$	spherical- 10 μm radius	1) none 2) -500 MPa
$E=72\text{ GPa}$ $\sigma_y=3\text{ GPa}$	conical- 70.3° half-angle	3) none 4) -500 MPa
$E=72\text{ GPa}$ $\sigma_y=600\text{ MPa}$	spherical- 10 μm radius	5) none 6) -500MPa
$E=72\text{ GPa}$ $\sigma_y=600\text{ MPa}$	conical- 70.3° half-angle	5) none 6) -500MPa

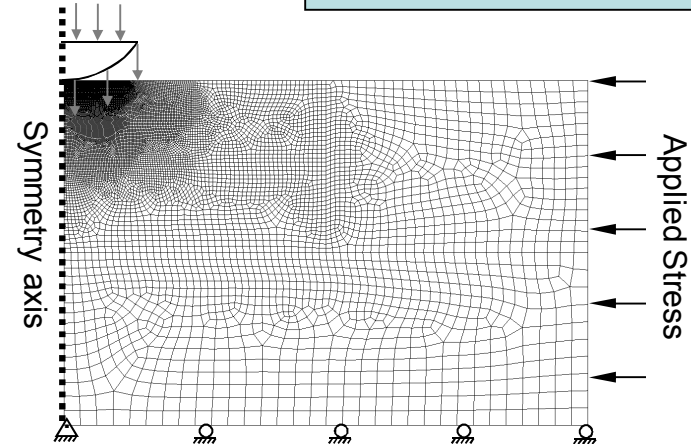
Simulation Details:

- frictionless
- *rigid tip*
- substrate 30x40 μm –fixed
- approx. 20,000 elements
- axisymmetric elements

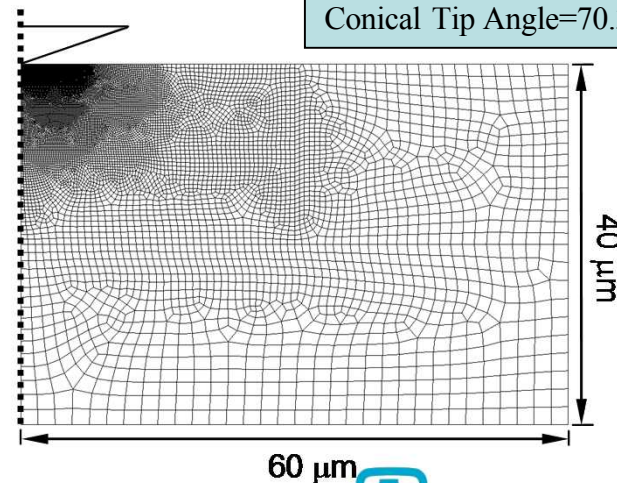
- FE meshes and boundary conditions

1 μm displacement
boundary condition

Spherical Tip Radius=10 μm



Conical Tip Angle=70.3°

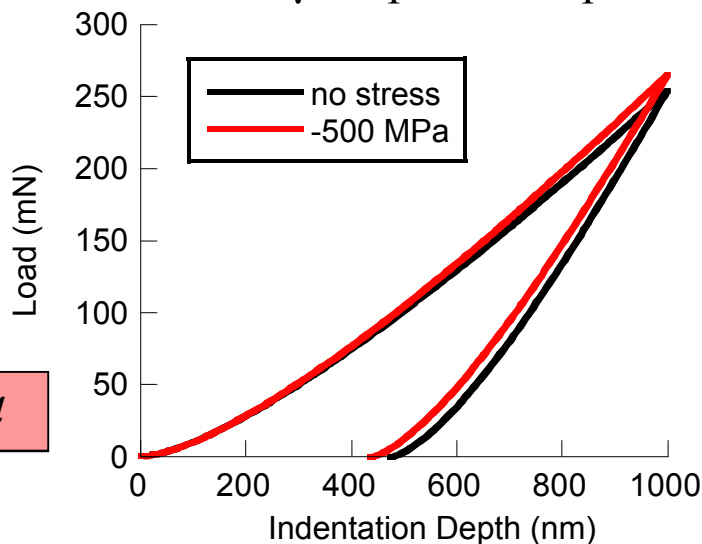


Stressed substrates gave a significantly different load-displacement response in every simulated case

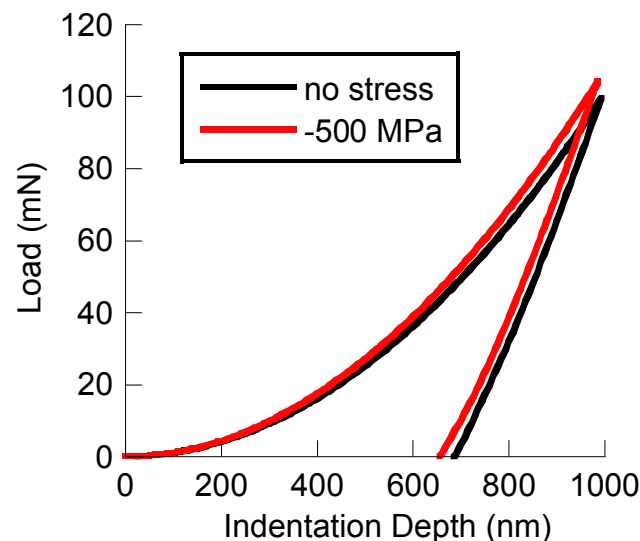
$E = 72 \text{ GPa}$
 $\sigma_y = 3 \text{ GPa}$

$$E/\sigma_y = 24$$

10 μm spherical tip

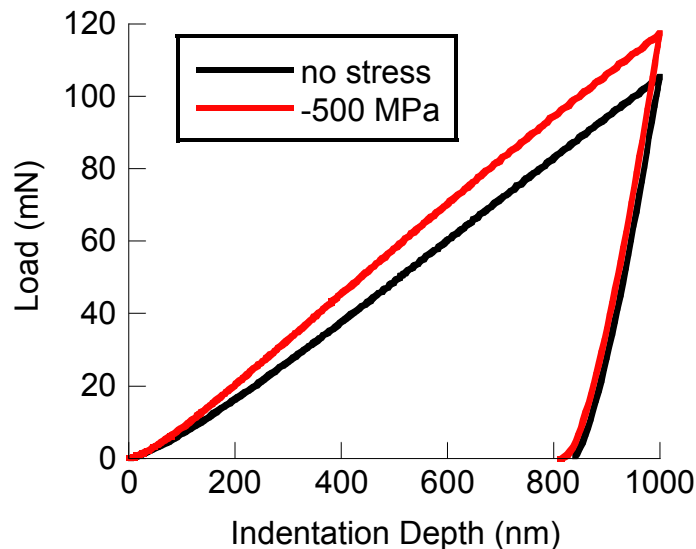
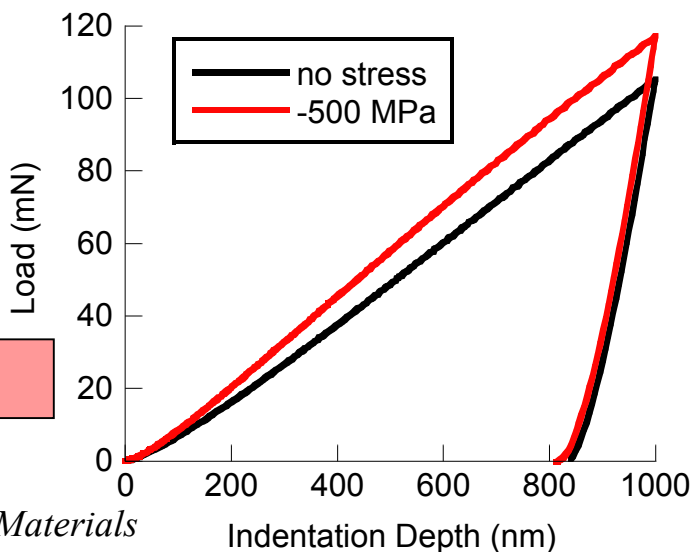


conical tip



$E = 72 \text{ GPa}$
 $\sigma_y = 600 \text{ MPa}$

$$E/\sigma_y = 120$$



Tabular analysis of FE results begins to reveal trends:

E/ σ_y Ratio (E=72 GPa)	Tip Geometry	Substrate Stress (MPa)	β *Modulus O-P method (GPa)	β *Modulus True A_c (GPa)	Hardness O-P Method (GPa)	Hardness True A_c (GPa)
24	Spherical	0	83	82	6.04	6.04
($\nu=0.2$)	10 μm rad.	-500	81	81	6.49	6.45
24	Conical	0	81	79	6.48	6.22
($\nu=0.2$)	70.3° half-angle	-500	83	80	7.05	6.56
120	Spherical	0	93	84	1.97	1.59
($\nu=0.3$)	10 μm rad.	-500	98	83	2.17	1.55
120	Conical	0	95	80	2.19	1.56
($\nu=0.3$)	70.3° half-angle	-500	103	81	2.50	1.56

β is a correction factor dependent on tip geometry ≈ 1.04 -1.1

- O-P method** relies on unloading portion of indentation load-displacement curve for computing contact stiffness, S , contact area, A_c , and ultimately hardness, H , and modulus, E_r , material properties

O-P method does not account for indentation pile-up

Key Formulas →

$$S = \frac{2\beta}{\sqrt{\pi}} E_r \sqrt{A_c}$$

$$H = P / A$$

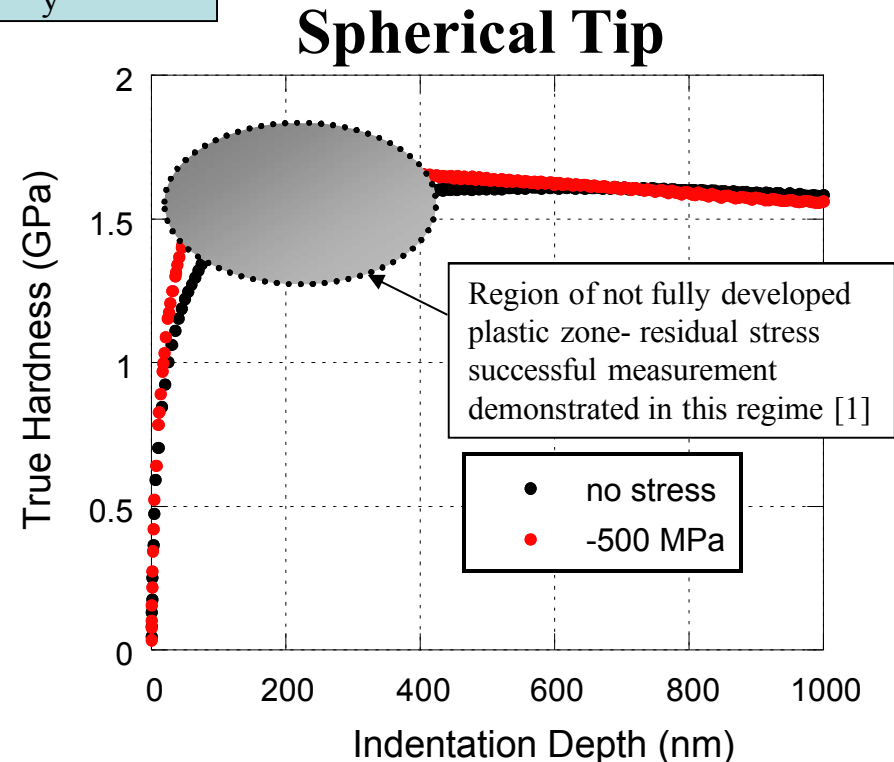
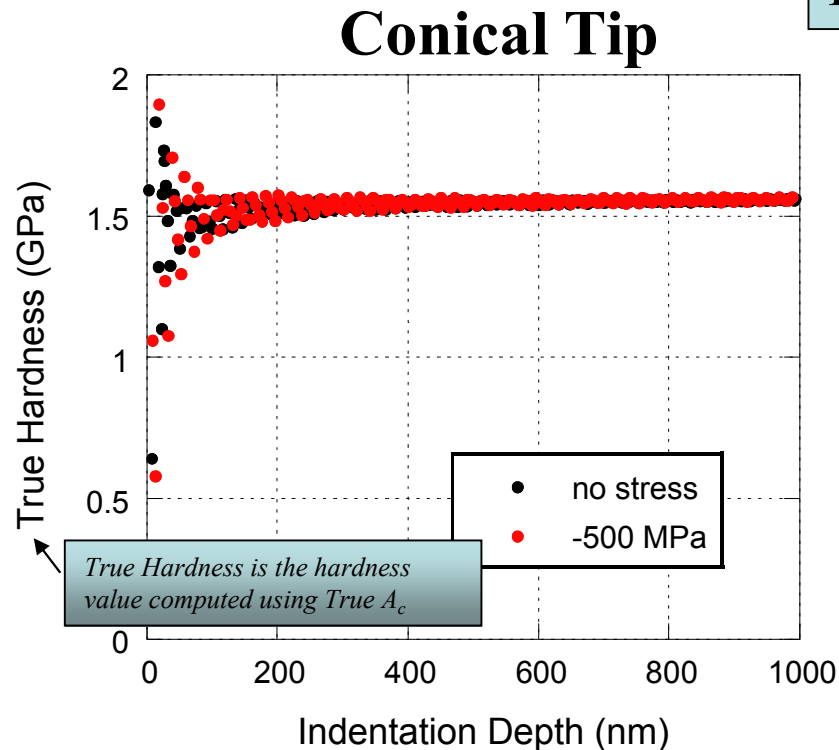
Oliver, W.C., and Pharr, G.C., "An improved technique for determining hardness and elastic modulus using load and displacement sensing indentation experiments" J. Mater. Res., Vol. 7, No. 6, 1992, p 1564.

- Properties determined using **True A_c** rely on the contact area between the indenter tip and substrate material determined by the finite element simulation

Using True A_c determined by FE accounts for indentation pile-up

Using true contact area in hardness determination removes the influence of indentation pile-up

$$E/\sigma_y = 120$$

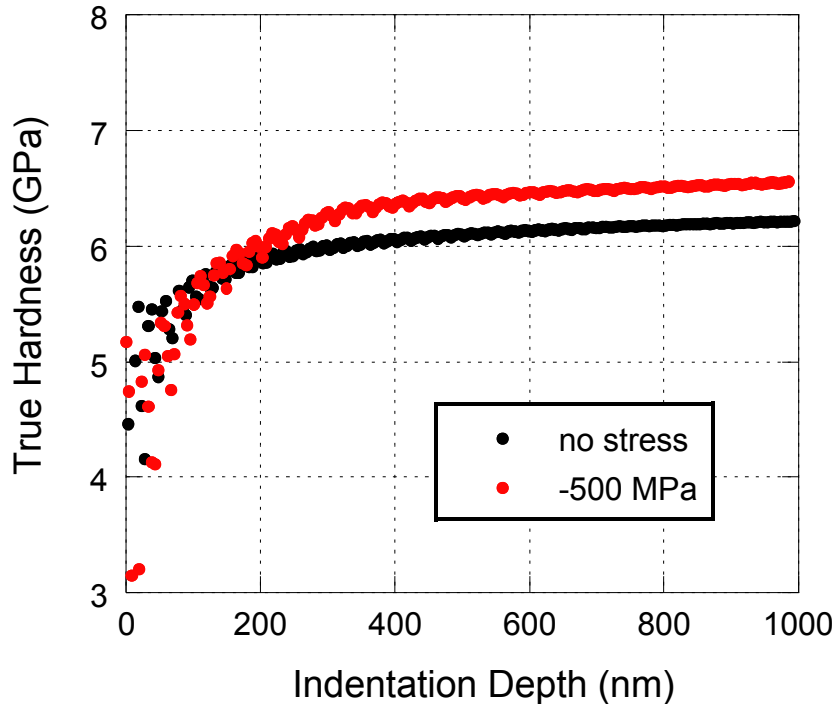


- Compressive residual stress does not impact the true hardness when a material with a high E/σ_y ratio is indented with a conical tip
- Compressive residual stress measurably impacts the true hardness when a material with a high E/σ_y ratio is indented with a spherical tip *only over a certain range of indentation depths*

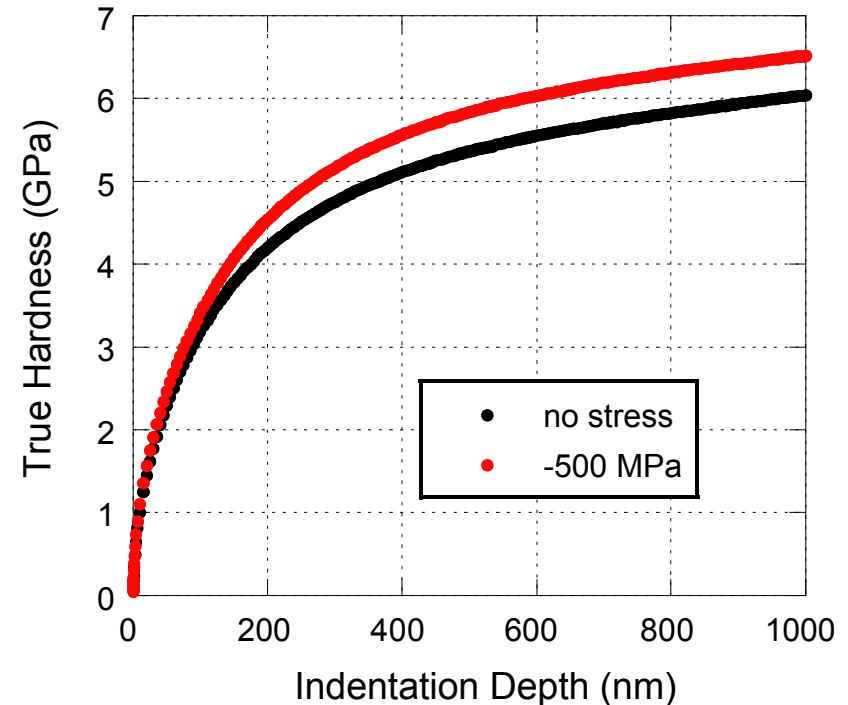
Measurable difference in instrumented indentation response in materials with low E/σ_y ratio

$$E/\sigma_y = 24$$

Conical Tip



Spherical Tip

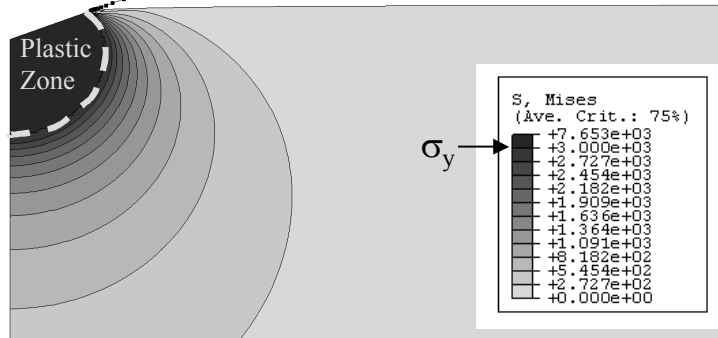


- Compressive residual stress measurably impacts the true hardness when a material with a low E/σ_y ratio is indented with either a conical or spherical tip across a wide range of indentation depths, *why?*

Stress distributions at 1 μ m indentation depth conical tip

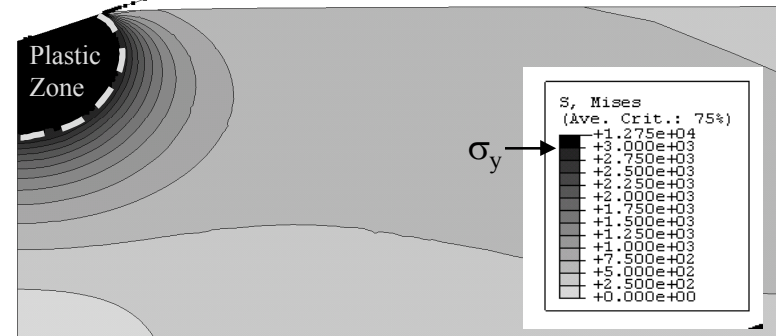
$$E/\sigma_y = 24$$

substrate stress = 0 MPa



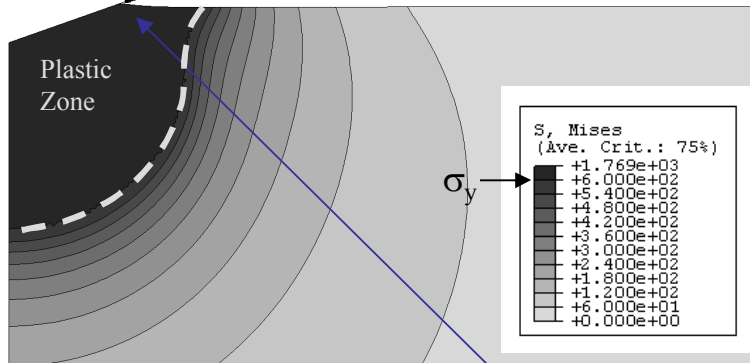
$$E/\sigma_y = 24$$

substrate stress = -500 MPa



$$E/\sigma_y = 120$$

substrate stress = 0 MPa



$$E/\sigma_y = 120$$

substrate stress = -500 MPa

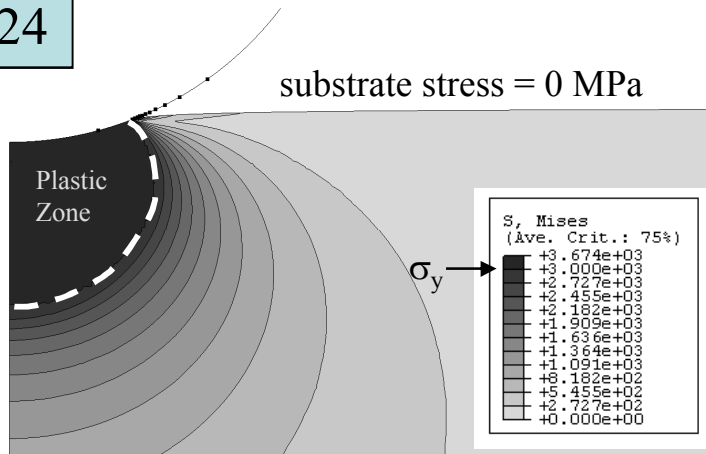


Indentation Pile-up

Stress distributions at 1 μ m indentation depth spherical tip

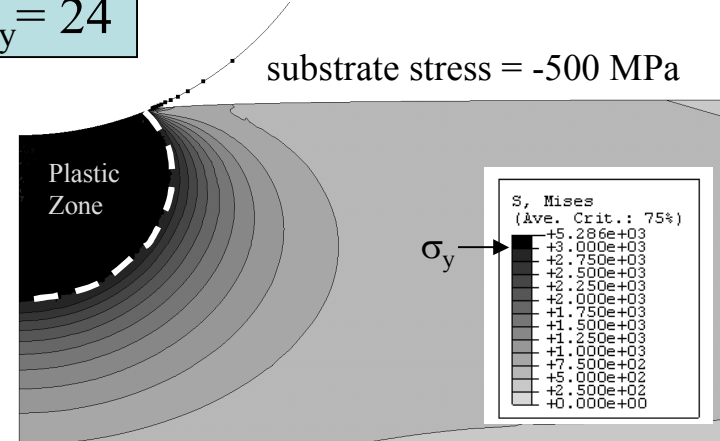
$$E/\sigma_y = 24$$

substrate stress = 0 MPa

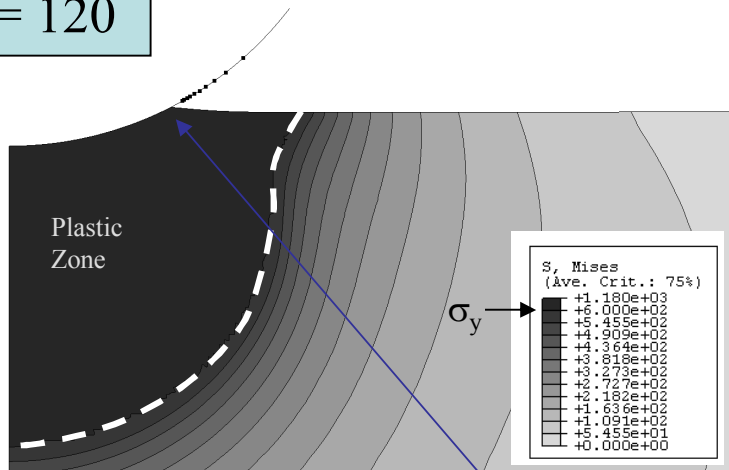


$$E/\sigma_y = 24$$

substrate stress = -500 MPa

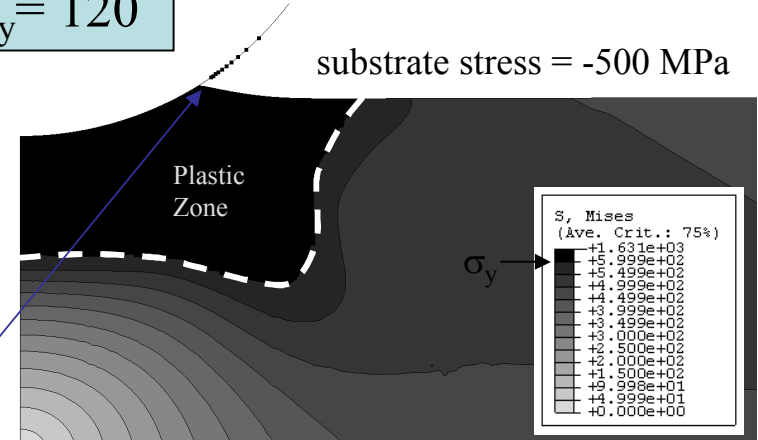


$$E/\sigma_y = 120$$

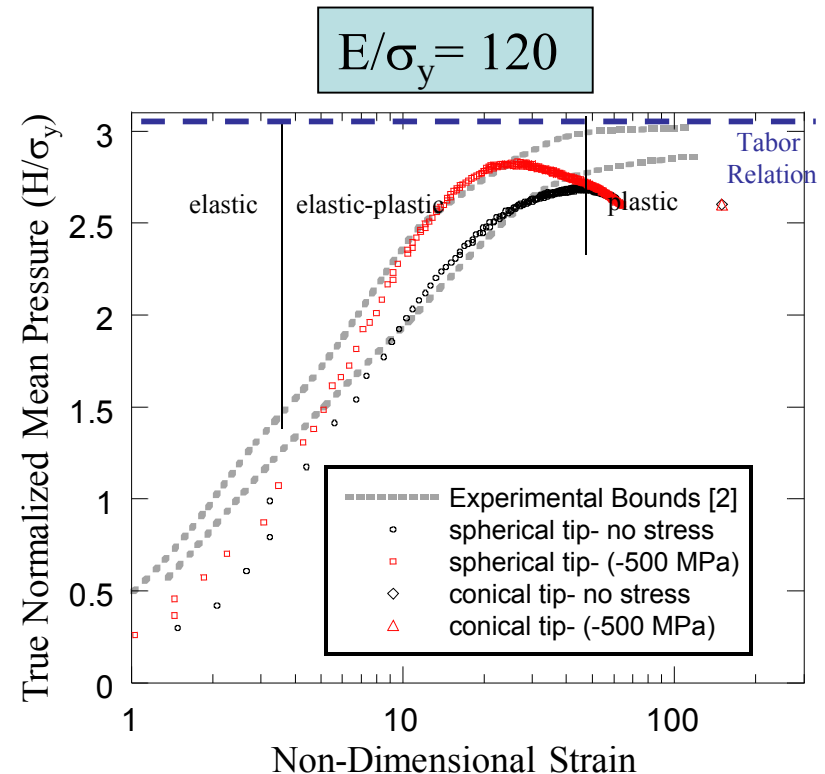
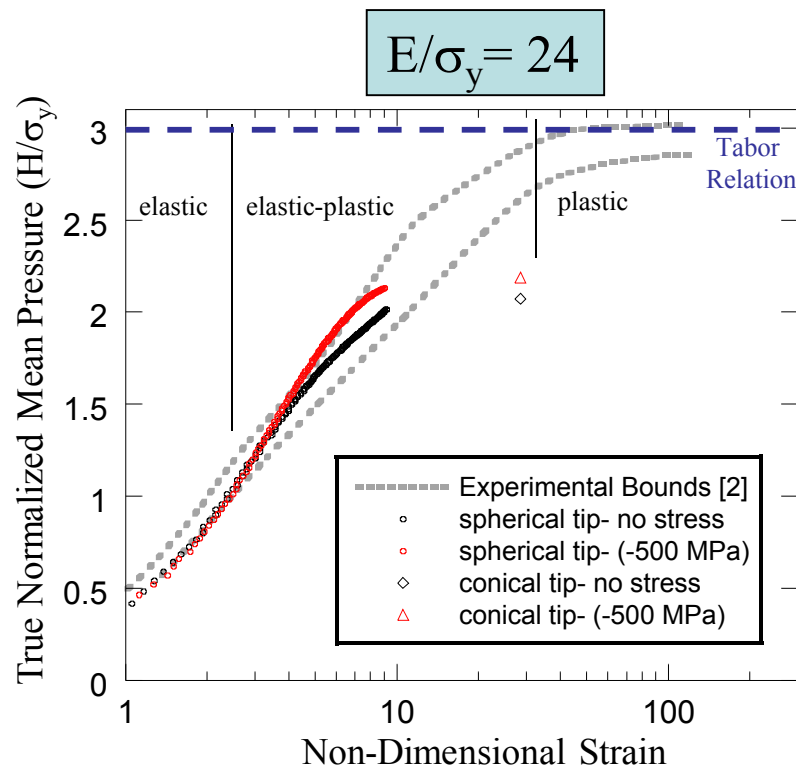


$$E/\sigma_y = 120$$

substrate stress = -500 MPa



When measuring residual stress, the "Universal Curve" shows the benefit of performing experiments in the elastic-plastic regime



Non-Dimensional Strain

$$\left(\frac{E_r a}{\sigma_y r} \right) \quad \text{for spherical tip,} \\ (a/r) \text{ is depth to tip radius ratio}$$

$$\left(\frac{E_r \tan \beta}{\sigma_y} \right) \quad \text{for conical tip}$$

- plastic refers to "fully developed" plastic zone underneath tip. In the absence of indentation pile-up, this result demonstrates that distinguishing between stress and unstressed material using nanoindentation is not possible in the plastic region of the universal curve.
- A fully developed plastic zone may never be achieved during indentation of a low E/σ_y material



Observations and conclusions drawn from simulated results

- A plastic zone always constrained to region underneath tip, as demonstrated in the low E/σ_y simulated results corresponds to a not fully developed plastic zone.

These results showed a measurable difference in indentation response between stressed and unstressed substrates for both spherical and pyramidal tip geometries

- A plastic zone that breaks out to the substrate surface, as demonstrated in the high E/σ_y simulated results corresponds to a fully developed plastic zone.

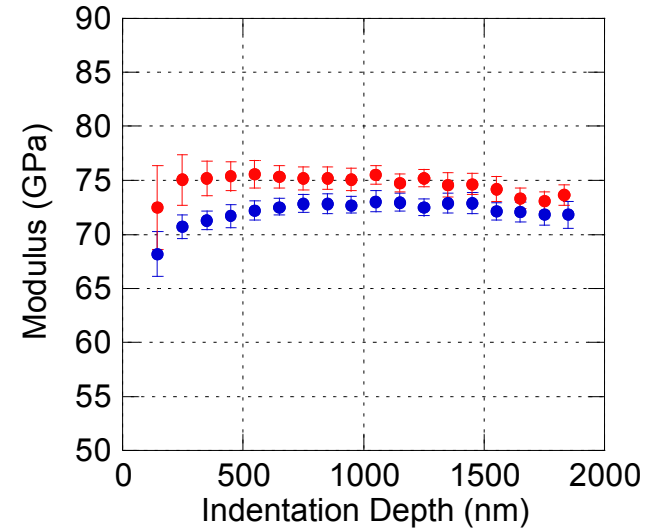
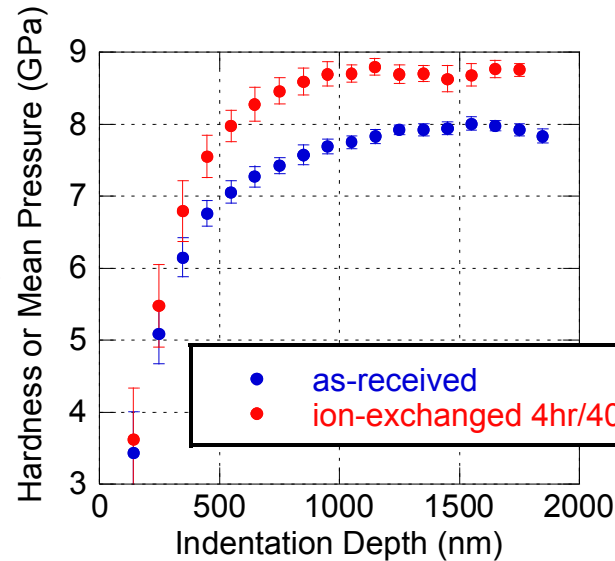
These results also showed a measurable difference in indentation response between stressed and unstressed substrates, that difference is completely attributable to "indentation pile-up" phenomenon

An indentation experiment that creates a large elastic zone and a small confined plastic zone is most useful for measuring influence of residual stress using nanoindentation. Glasses and Ceramics are favorable materials for this type of measurement because of their high E/σ_y ratio

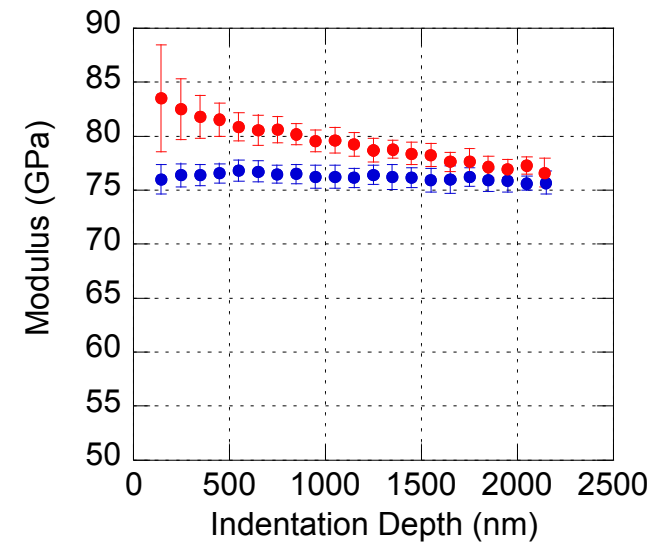
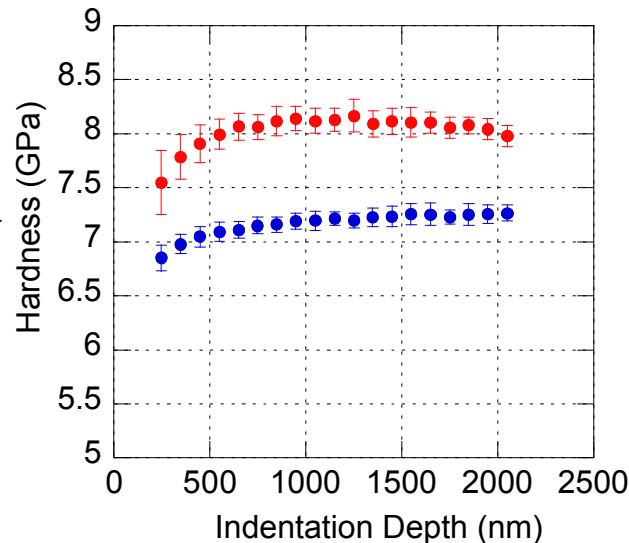
Instrumented indentation experiments showing a measurable difference between stressed and unstressed glass

Nanoindentation
results on PPG
Glass-500 mN load

10 μm spherical tip



berkovich tip





Conclusions

