

Dislocation kink-pair theory as a basis for temperature- and rate-dependent BCC crystal plasticity model

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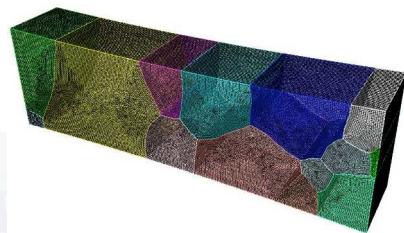
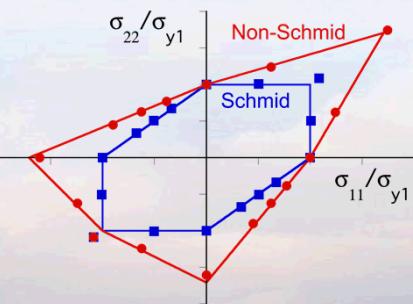
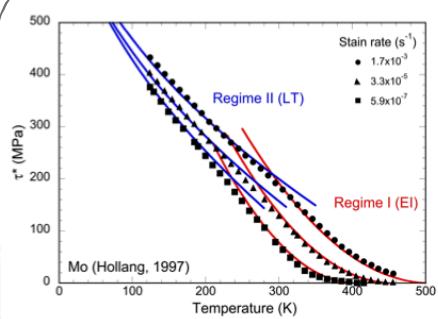
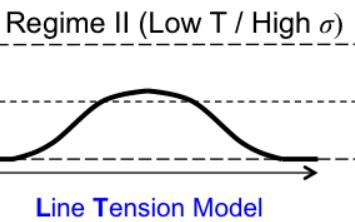
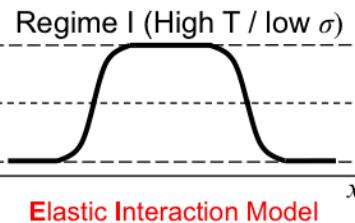
Multi-Scale Modeling of BCC Plasticity

Atomic Scale

Crystal Plasticity

Microstructure

Macro-Scale

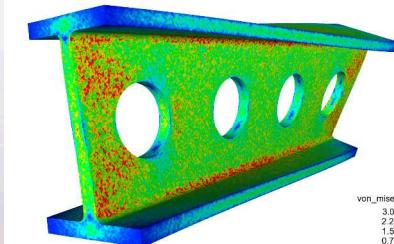
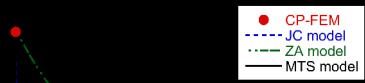


Dislocation Physics

Single Crystal Behavior

Polycrystal Properties

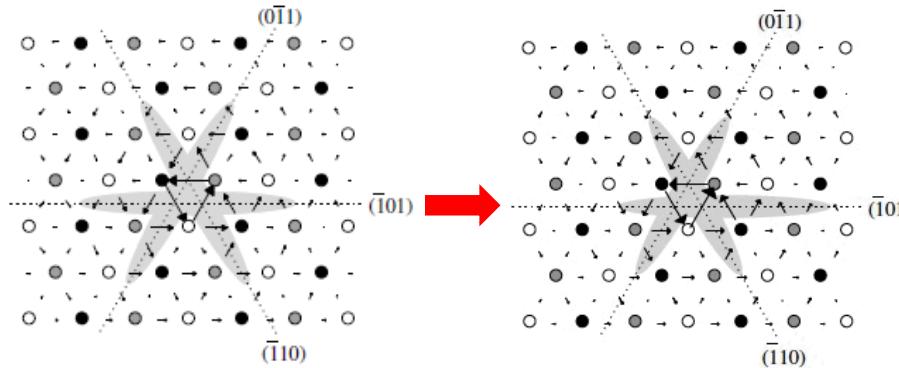
Macroscale Models



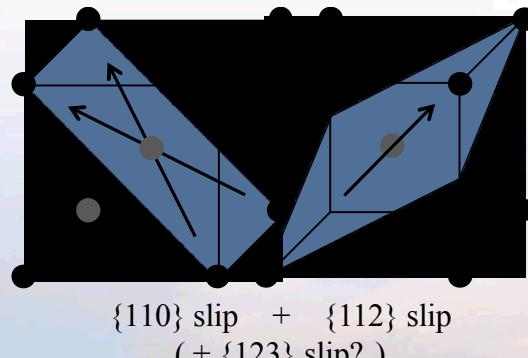
Nano: Atomic Properties of Dislocations in BCC Metals

Screw dislocations in BCC metals have complex atomic structures ...

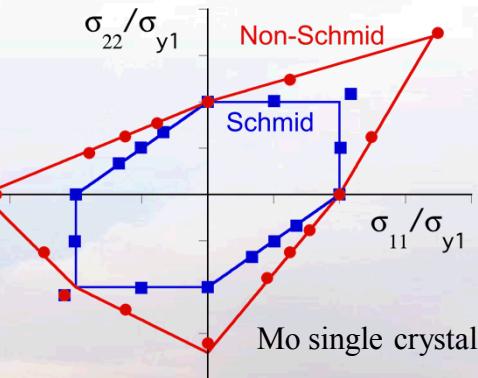
[111] zone depiction of a relaxed screw dislocation core in Mo



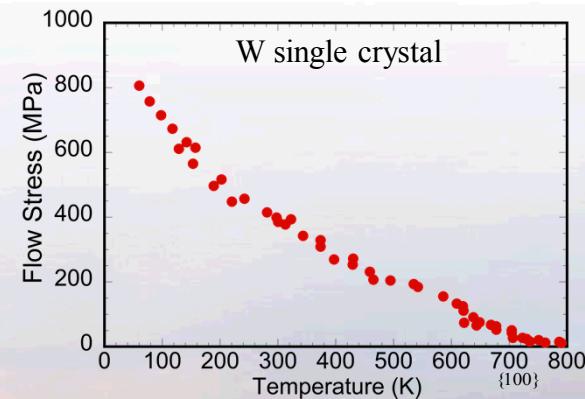
Distortion of the dislocation core under an applied shear stress



Uncertainty about important slip systems



Non-Schmid plasticity

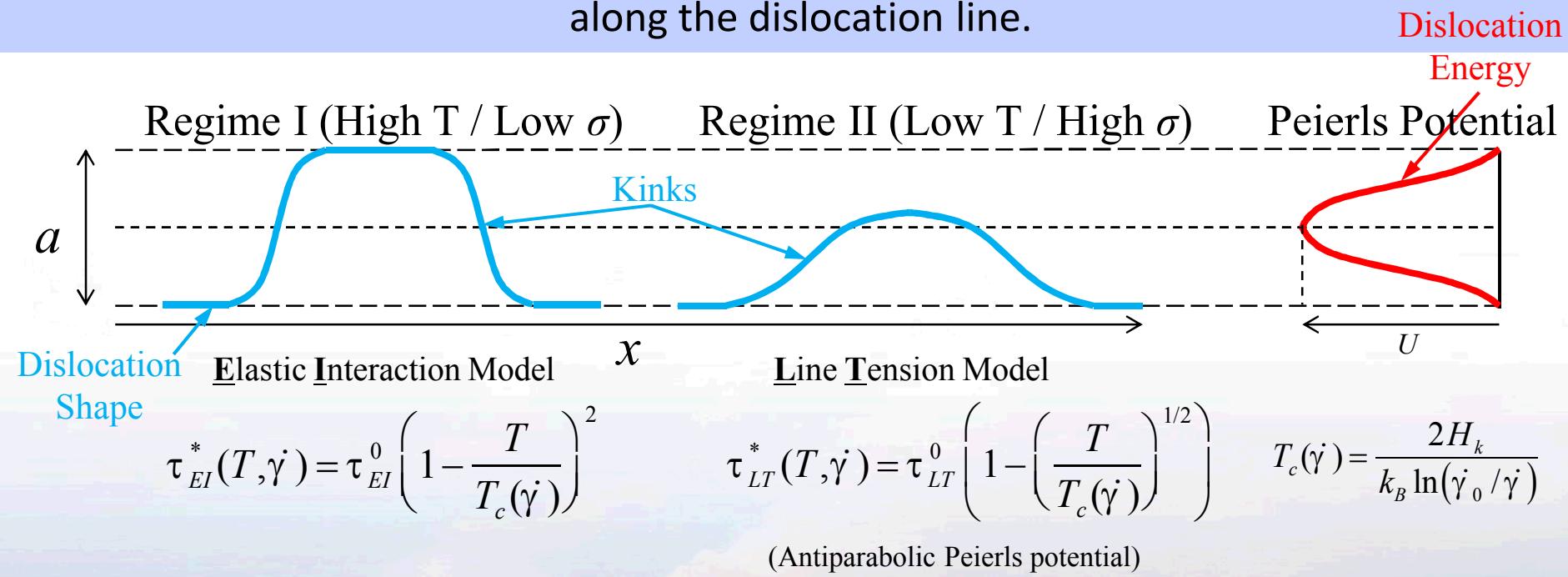


Strong temperature and strain rate sensitivity

... and this leads to various non-ideal behaviors and properties.

Dislocation Kink-Pair Theory

Dislocation motion occurs by the formation and propagation of atomic-scale kinks along the dislocation line.



The atomic-scale properties of BCC screw dislocations guide the development of a dislocation kink-pair model for how the temperature and strain-rate affect a BCC metal's yield stress.

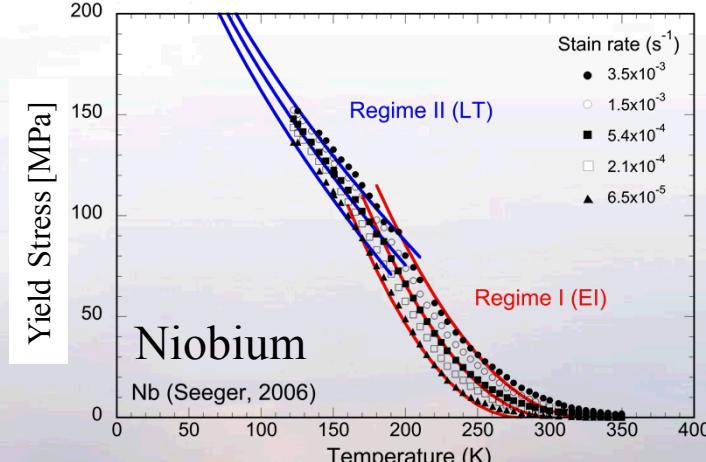
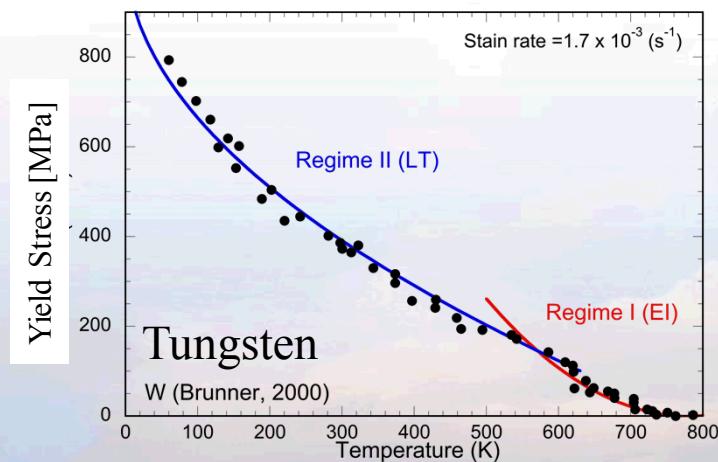
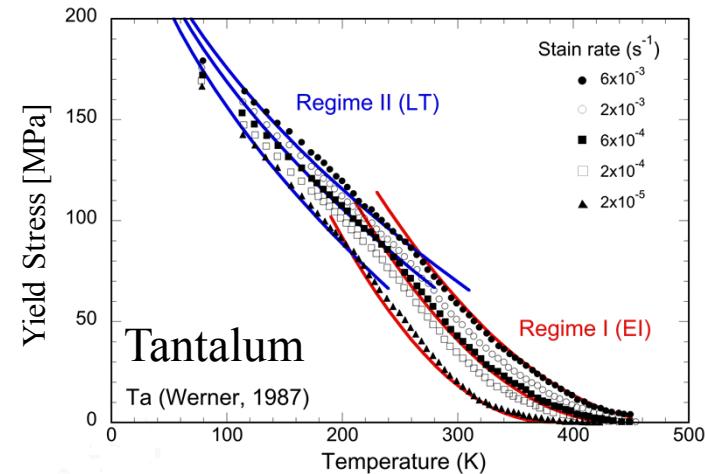
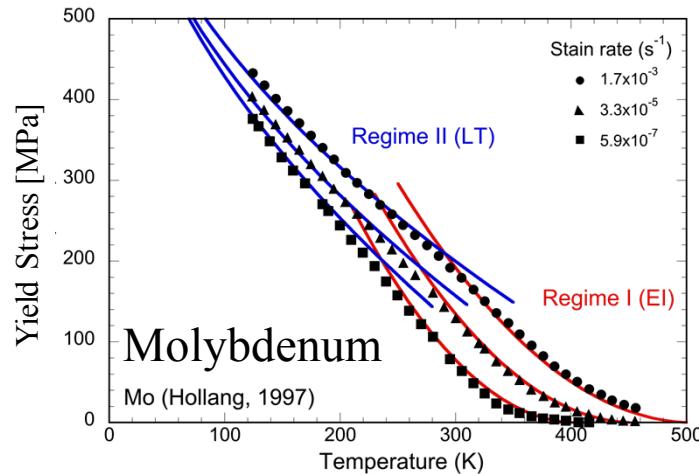
Seeger, Z. Metallk. 1981

Seeger, Mater. Sci. Eng. A, 2001

Butt, Phil. Mag. 2007

Argon, Strengthening mechanisms in crystal plasticity 2008

Confirming the form of the “Kink-Pair” Model and calibrating to single crystal data

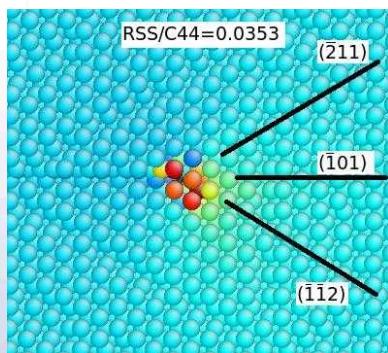


Our dislocation kink-pair model accurately reproduces experimental data for the yield stress over a range of BCC metals, temperatures, and strain rates.

Crystal Plasticity Modeling discretely models each grain in the material with its elastic and plastic anisotropy.

- Crystal plasticity = Grain-level (mesoscale) approach to materials modeling using multiscale strategies
- Explicitly model discrete grains and slip systems (anisotropy, texture evolution,...)

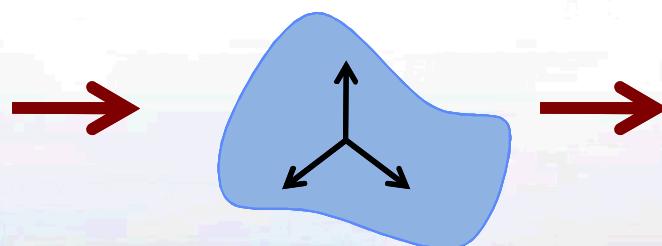
Atomic phenomenology:
Fundamental deformation mechanisms



Yield criterion

$$\begin{aligned}\sigma_{cr}^{app} [a_0 \mathbf{m}^{(s)} \mathbf{n}^{(s)} + a_1 \mathbf{m}^{(s)} \mathbf{n}^{(s')} \\ + a_2 (\mathbf{n}^{(s)} \times \mathbf{m}^{(s)}) \mathbf{n}^{(s)} \\ + a_3 (\mathbf{n}^{(s)} \times \mathbf{m}^{(s)}) \mathbf{n}^{(s')}] = \tau_{cr}\end{aligned}$$

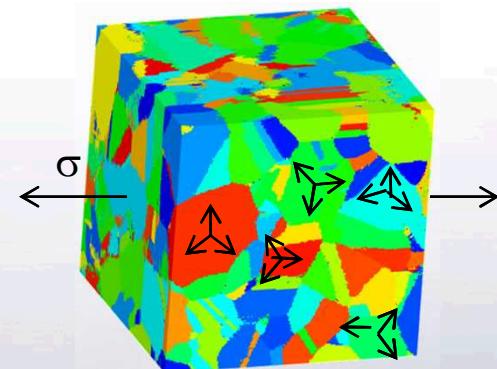
Single crystal plasticity:
Deformation of one, isolated crystal



Constitutive law

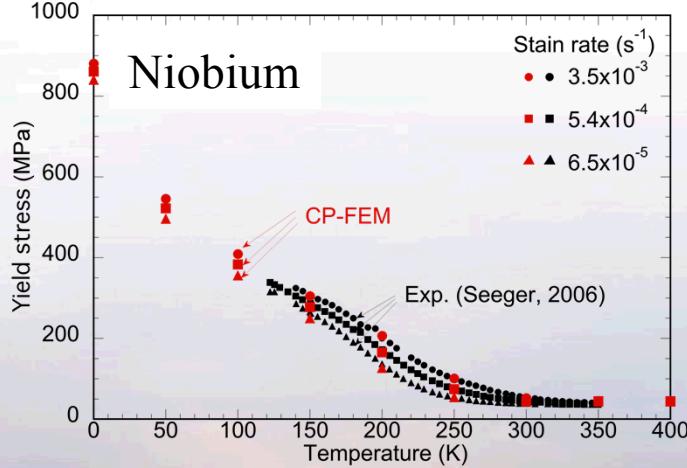
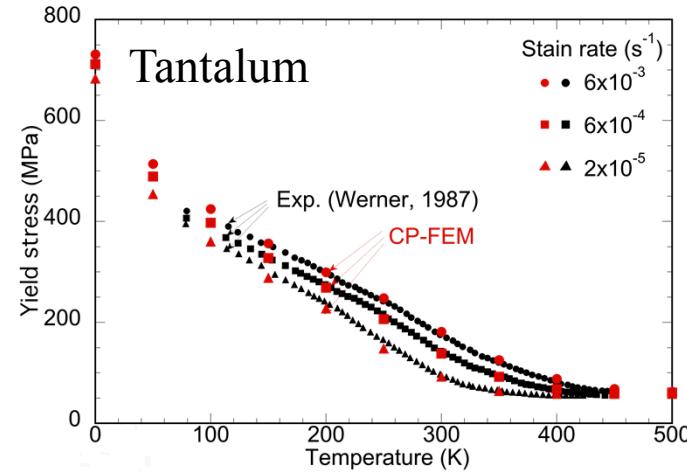
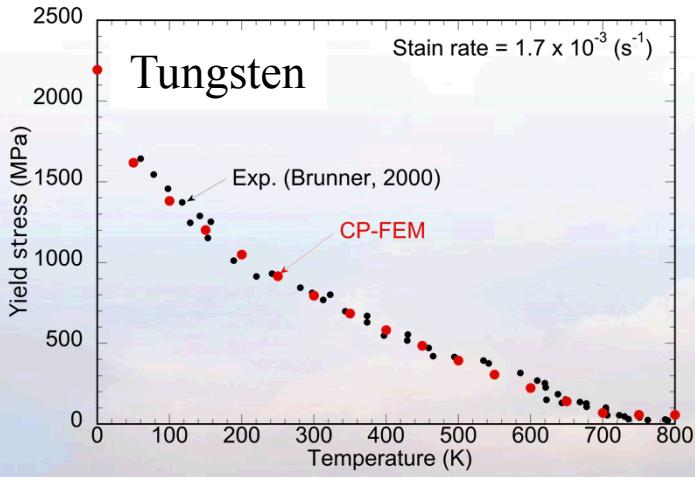
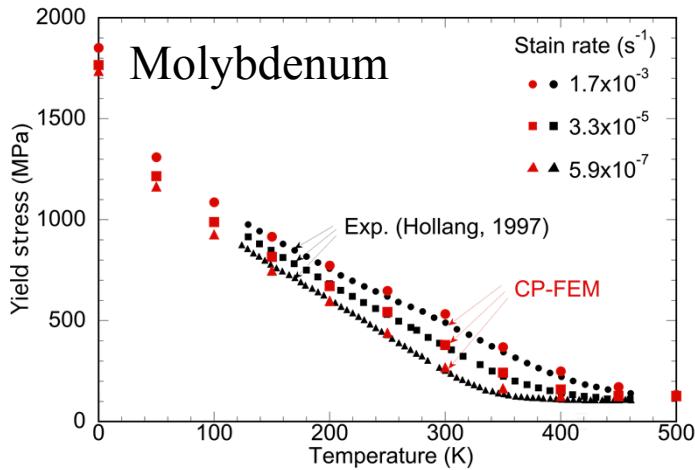
$$\dot{\gamma}^{(s)} = \frac{\tau^{(s)}}{\tau_{cr}} \left| \frac{\tau^{(s)}}{\tau_{cr}} \right|^{m-1}$$

Polycrystal plasticity:
Assemble single crystals into polycrystalline ensemble



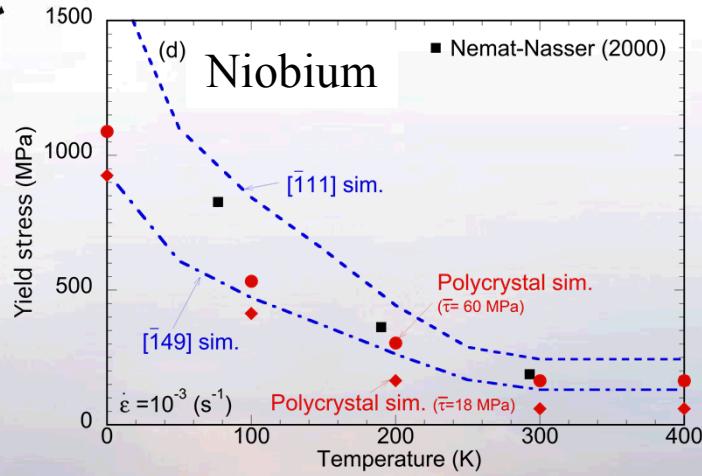
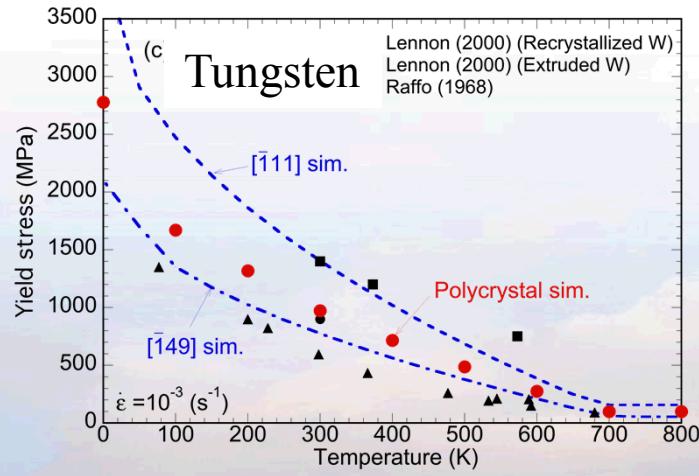
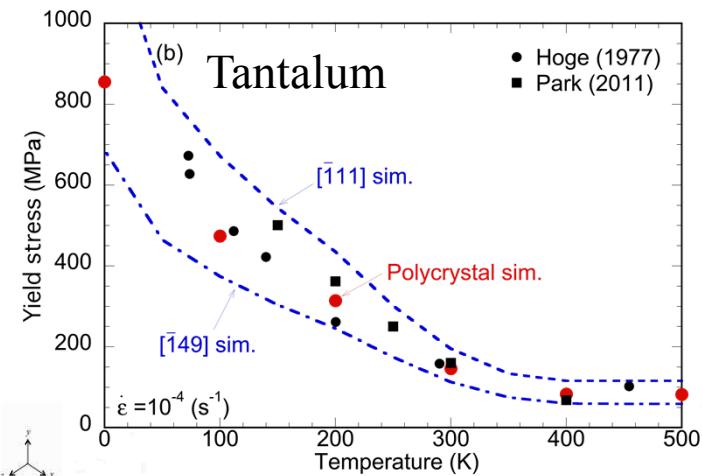
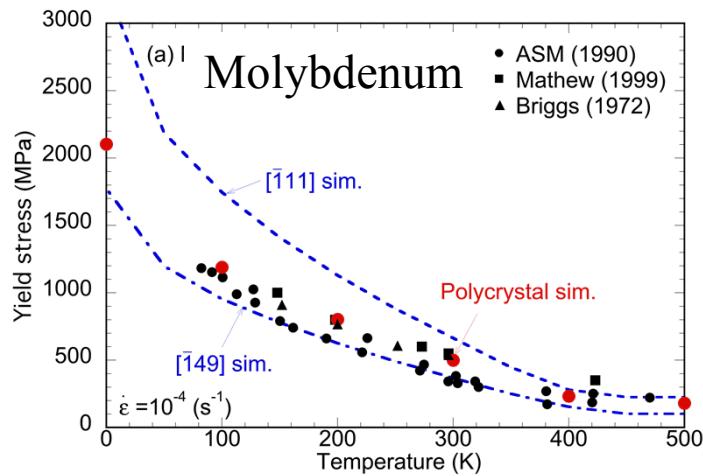
Prediction of collective deformation behavior

Micro: Implementation into Single-Crystal Plasticity



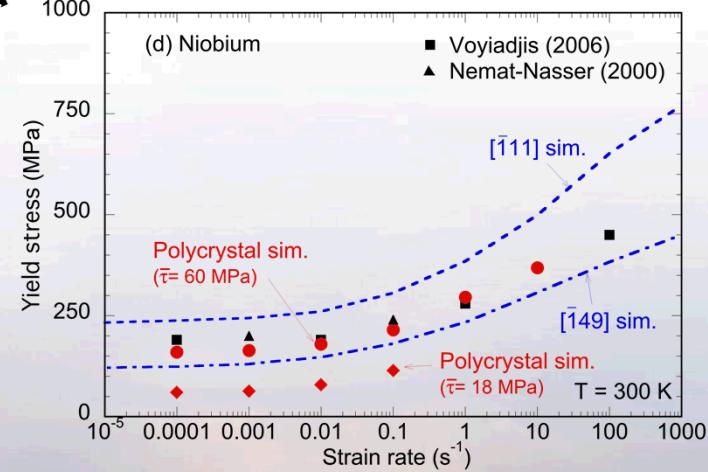
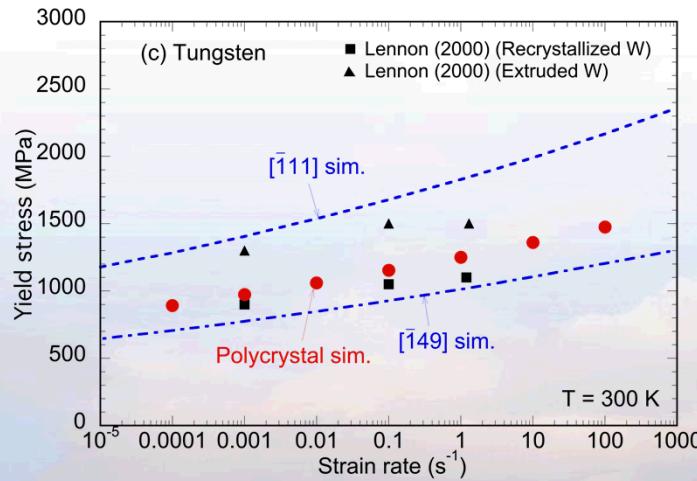
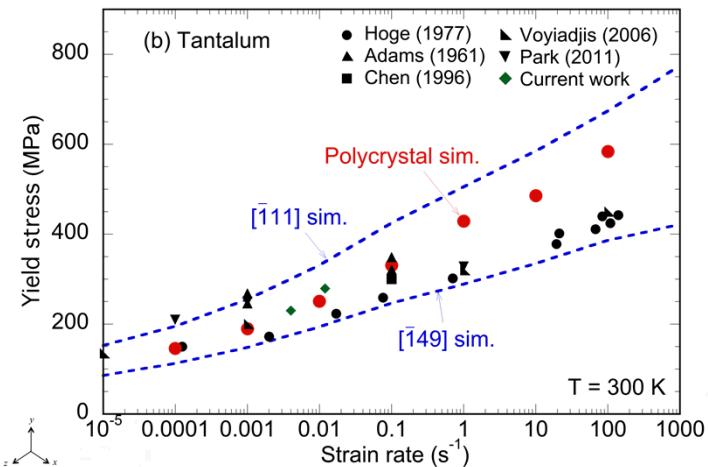
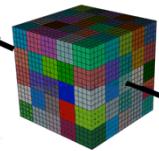
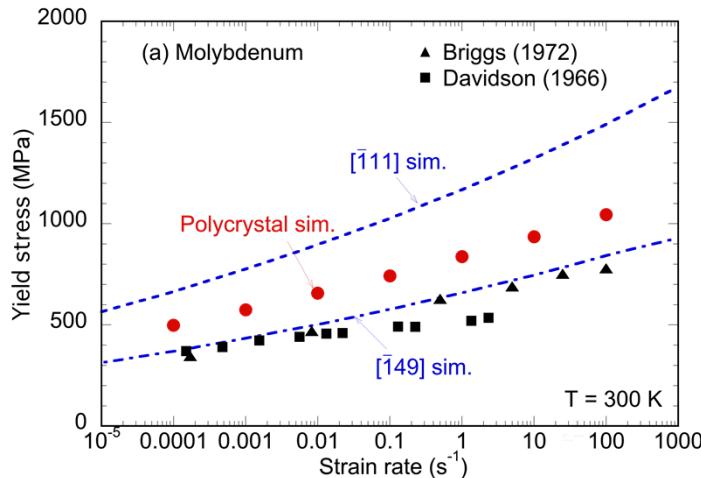
Moving up in scale and implementing the theory into a crystal plasticity finite element model (CP-FEM) for single-crystal BCC plasticity, the quality of validation is retained.

Meso: Up-Scaling into Polycrystal Plasticity



The next length scale is the polycrystal, where CP-FEM simulations agree very well with experiments, even when calibrated only to independent, single-crystal tests.

Meso: Up-Scaling into Polycrystal Plasticity



Moving up again in scale, CP-FEM simulations agree very well with experiments, even when using only calibrations to independent, single-crystal tests.

Macro: Classical Continuum Models

- Johnson and Cook (JC) model (Johnson and Cook, 1983, 1985)

$$\sigma_y^{JC} = A(1 + C \ln \dot{\varepsilon})(1 - T^{*m})$$

- Zerilli-Armstrong (ZA) model (Zerilli and Armstrong, 1987)

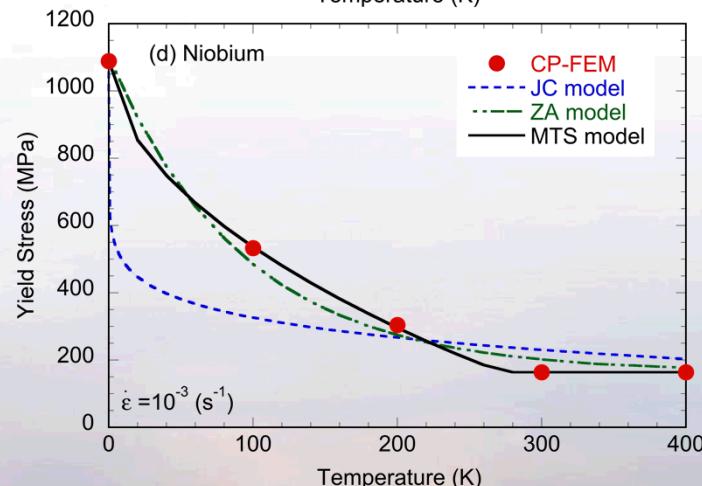
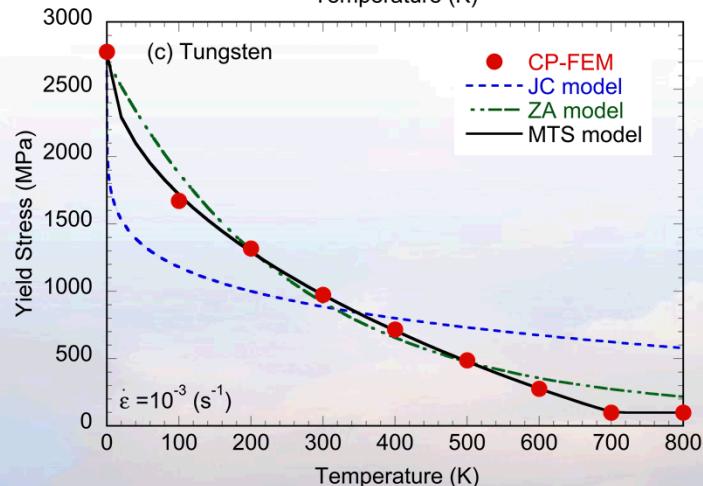
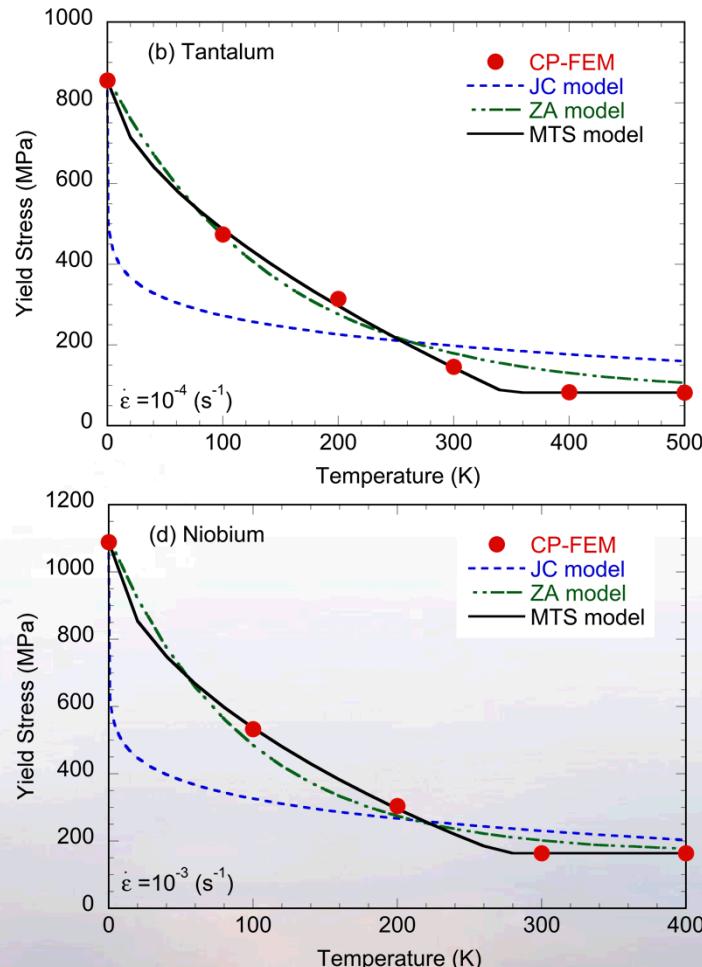
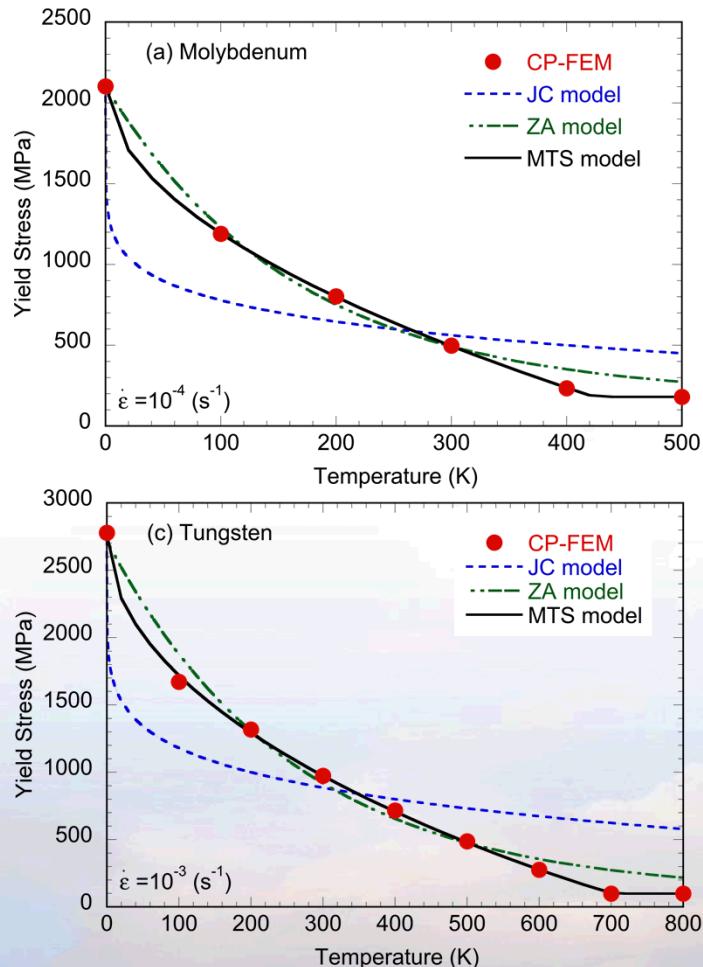
$$\sigma_y^{ZA} = C_0 + C_1 \exp(-C_3 T + C_4 T \ln \dot{\varepsilon})$$

- Mechanical Threshold Stress (MTS) model (Follansbee, 1988)

$$\sigma_y^{MTS} = \sigma_0 + \hat{\sigma} \left(1 - \left(\frac{k_B T}{G_0} \ln \frac{\dot{\varepsilon}}{\dot{\varepsilon}_0} \right)^{1/q} \right)^{1/p}$$

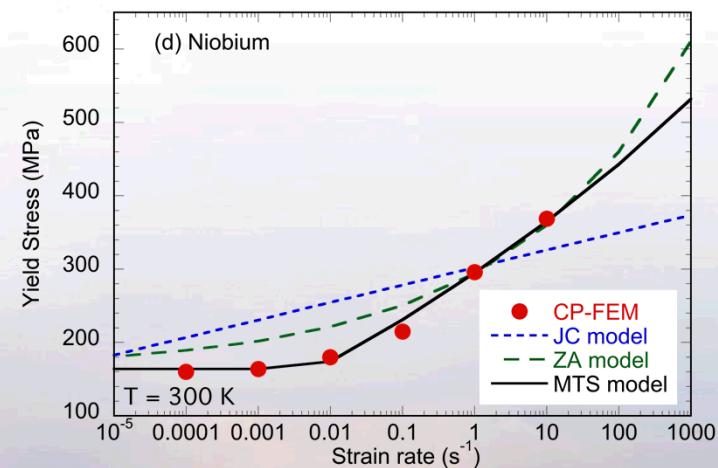
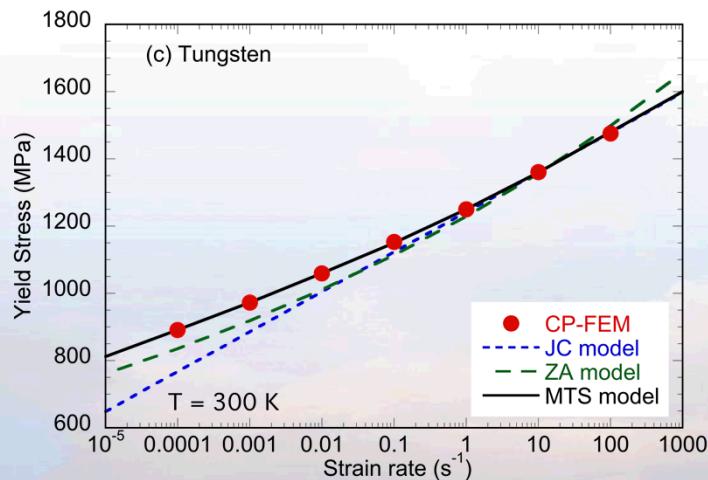
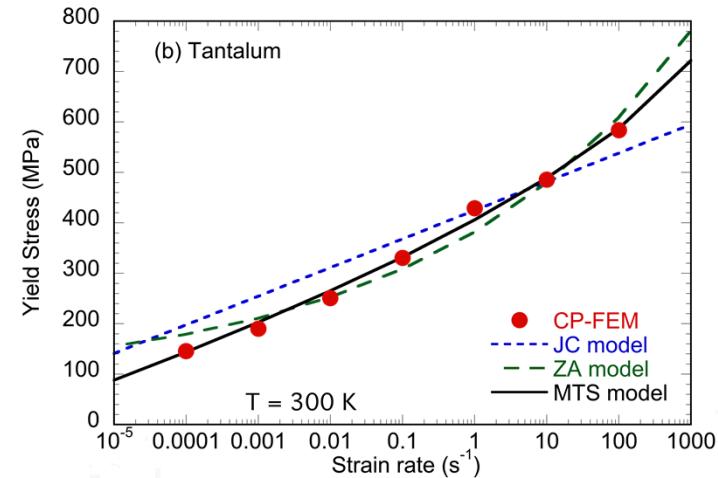
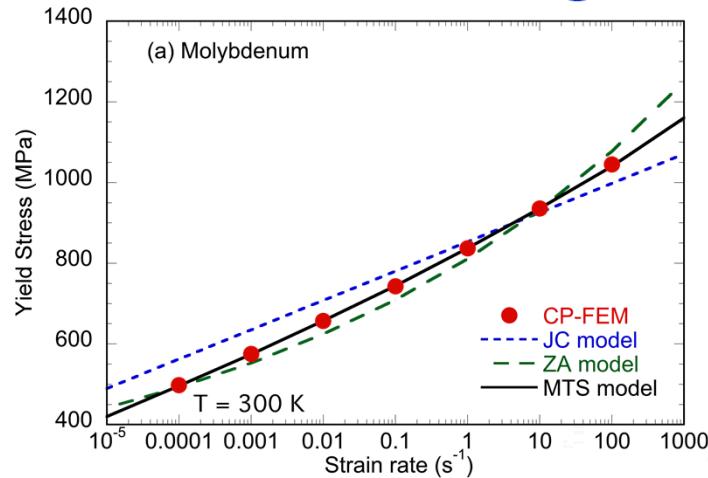
Sandia's macro-scale engineering analysis codes use conventional constitutive models to describe temperature and strain-rate effects.

Macro: Interfacing CP-FEM into Continuum Models



Our CP-FEM simulations, derived from atomic-scale considerations of BCC dislocations, provide a direct link to these continuum models' parameters.

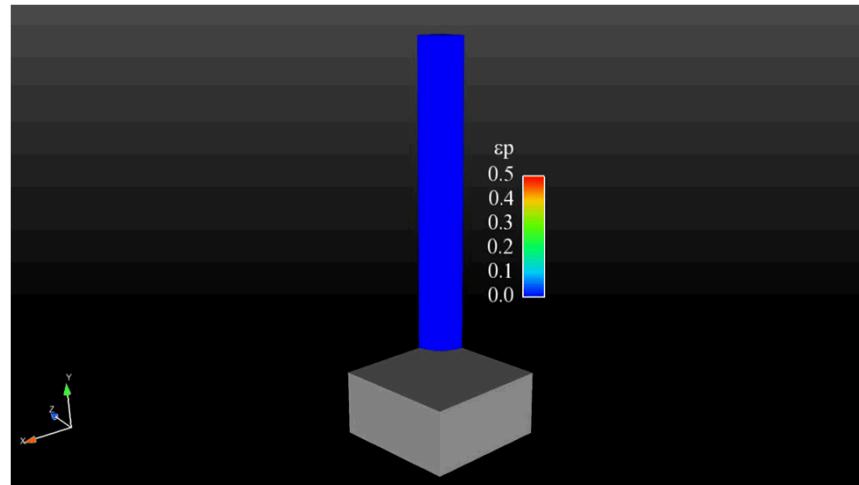
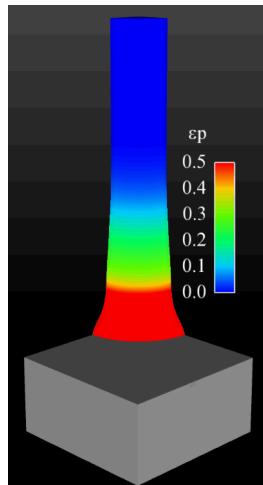
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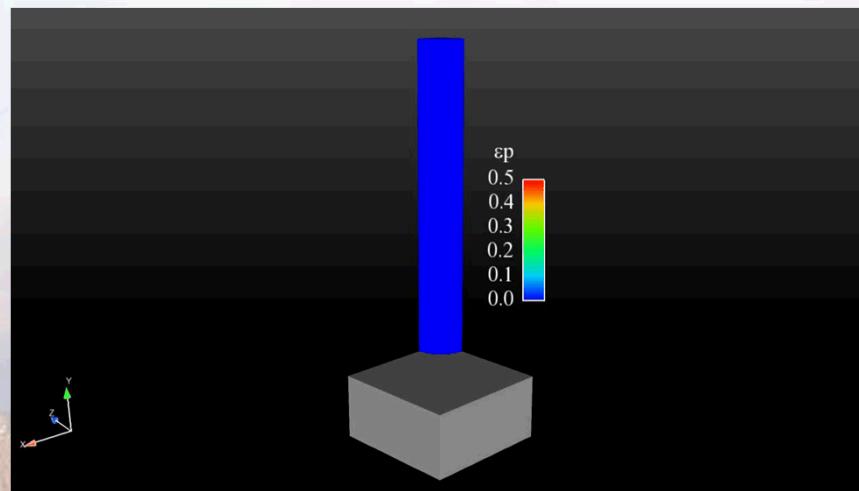
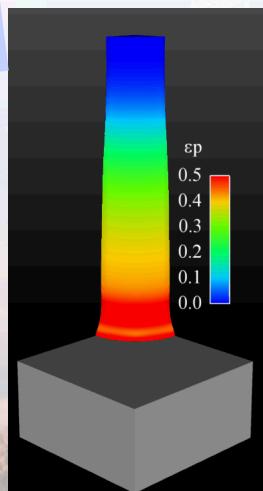
What's next: From Atomic-Scale Dislocation Theory into Alegra (Sandia parallelized continuum dynamics code)

Johnson-Cook



We are ramping up parameterizations of JC, ZA, MTS, and BCJ for use in codes such as Alegra and CTH

Zerilli-Armstrong



Summary

1. Existing dislocation kink-pair theory forms a basis for understanding how the peierls barrier manifests in temperature and rate-dependence
2. This theory provides single crystal flow rules that are now embedded in a polycrystal plasticity framework
3. The single crystal flow rules can be calibrated on existing single crystal literature data; the 4-parameter model form fits exceptionally well.
4. Within the crystal plasticity model, the form of the temperature and rate dependence matches experimental polycrystal data.
5. At the macro-scale, this kink-pair based crystal plasticity model compares very favorably with the MTS model for temperature strain-rate dependence.

Lim, Battaile, Carroll, Weinberger, Boyce, “A physically based temperature and strain rate dependent crystal plasticity model for BCC metals,” *J. Mechanics and Physics of Solids*, 2014.