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# In Situ Synthesis of Uranium Carbide and its High Temperature Cubic Phase

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[reiche@lanl.gov](mailto:reiche@lanl.gov), March 16, 2015

Lujan Neutron Scattering Center & Material Science & Technology Division



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# Motivation: New Ceramic Fuels



## New Fuels

- Uranium-carbides, -nitrides, -silicides, ... are actively researched.
- Phase diagram not fully established

## Accurate Models?

- Safety precautions based on accident scenario models
- Accurate model paramount

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# Motivation: $\text{UO}_x$ & $\text{UC}_x$ Nuclear Fuels

UC: higher thermal conductivity and higher fissile density than  $\text{UO}_2$

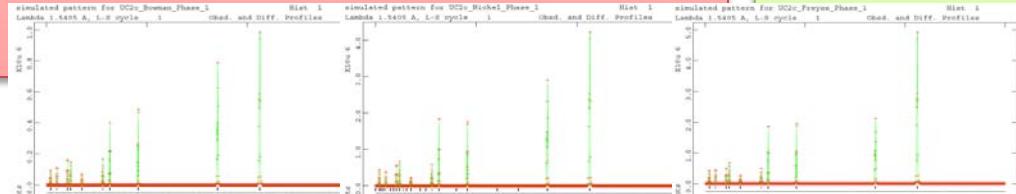
ICSD (*Inorganic Crystal Structure Database*) lists 15 entries for  $\text{UC}_2$ ;

- 12 describe tetragonal phase
- 3 non-quenchable cubic phase, stable 1769 - 2560°C; conflicting

Uranium powder is pyrophoric

Irradiation of actinides restrictive

XRD

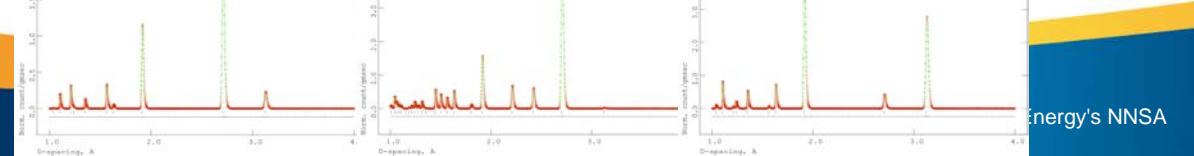


Bowman et al. ( $\text{NaCl}$ )

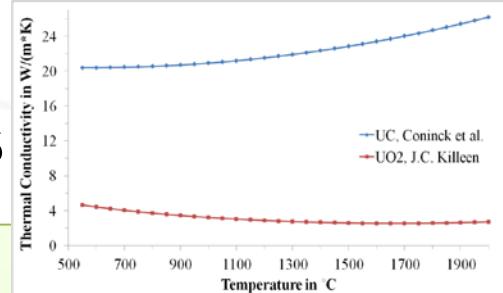
Bredig ( $\text{FeS}_2$ )

Wilson ( $\text{CaF}_2$ )

ND



Energy's NNSA



Study uranium carbides

Study  $\text{UC}_2$  in situ at  $T > 1769^\circ\text{C}$  using neutron diffraction

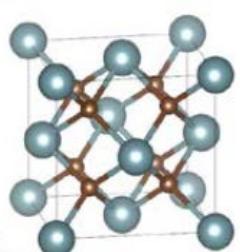


LANSCE, LANL

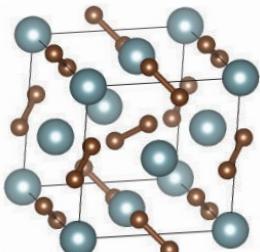
*If you can ship it, we can measure it*

# The U-C System

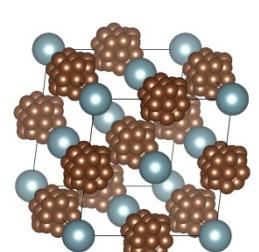
Phase	Structure Form	Space Group	Structure Type	Structure
$\delta$	$UC_x$	$Fm\bar{3}m$	NaCl	Face centered cubic
$\zeta$	$U_2C_3$	$I\bar{4}3d$	$Pu_2C_3$	Cubic
$\epsilon$	$UC_2$	$I4/mmm$	$CaC_2$	Body centered tetragonal



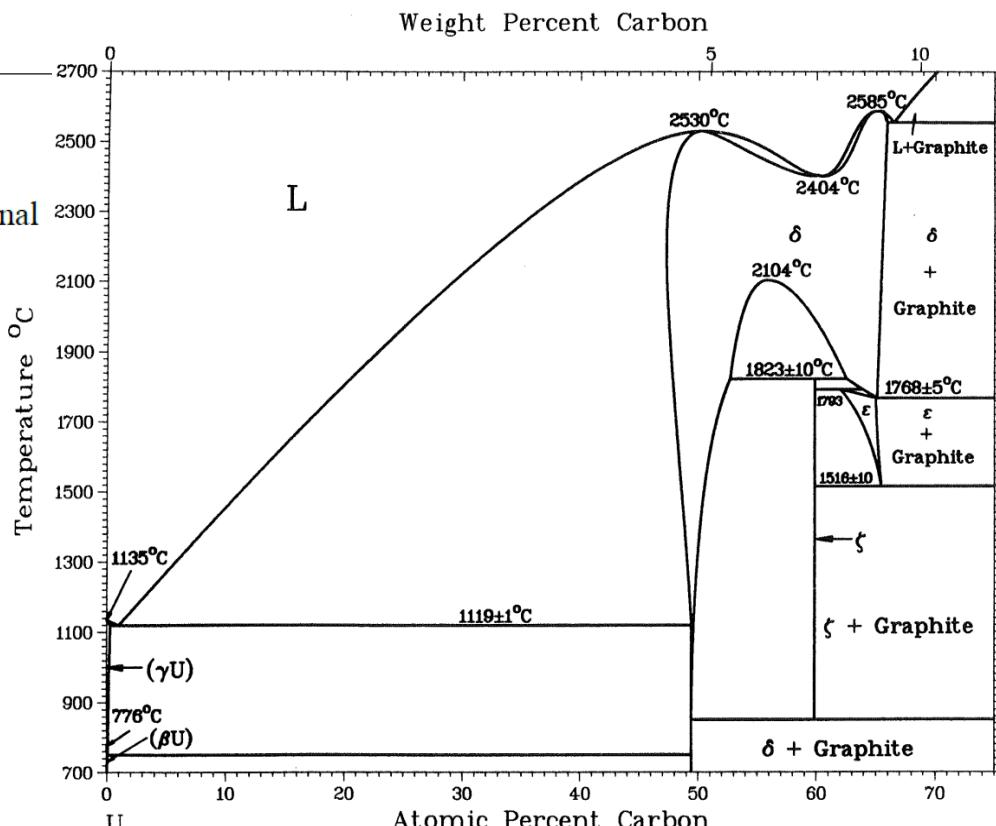
Wilson, W.B.  
1960  
 $CaF_2$



Bredig, M.A.  
1960  
 $FeS_2$



Bowman et al.  
1966  
 $NaCl$



In situ investigation required

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Assumes  $UC_2$  structure to be isotypic with  $CaF_2$   
according to W.B. Wilson, J. Am. Ceram. Soc. 43 (1960) 77.

[elsevier.nl/locate/jnucmat](http://elsevier.nl/locate/jnucmat)

# Thermodynamic modelling of the C–U and B–U binary systems

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Received 19 June 2000; accepted 7 November 2000

## Abstract

The thermodynamic modelling of the carbon–uranium (C–U) and boron–uranium (B–U) binary systems is being performed in the framework of the development of a thermodynamic database for nuclear materials, for increasing the basic knowledge of key phenomena which may occur in the event of a severe accident in a nuclear power plant. Applications are foreseen in the nuclear safety field to the physico-chemical interaction modelling, on the one hand the in-

PHYSICAL REVIEW B 81, 014101 (2010)

# First-principles study of uranium carbide: Accommodation of point defects and of helium, xenon, and oxygen impurities

Michel Freyss<sup>\*</sup>

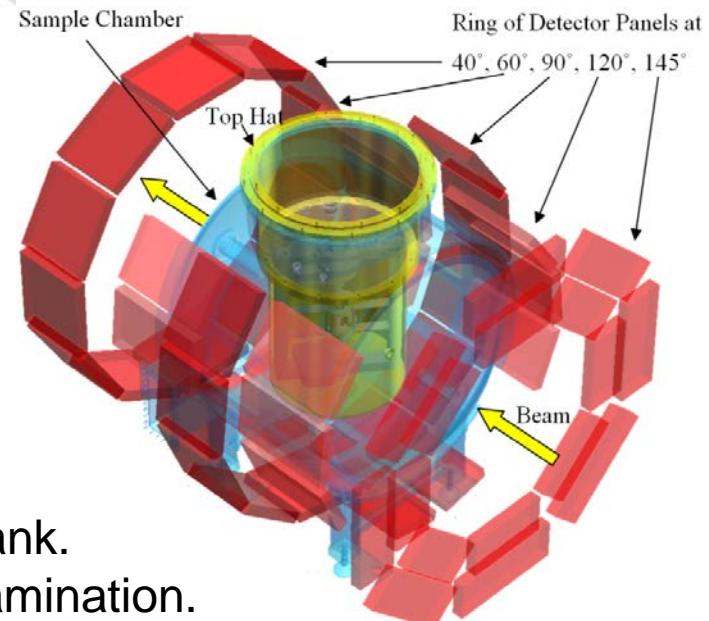
*CEA, DEN, Centre de Cadarache, DEC/SESC/LLCC, F-13108 Saint-Paul-lez-Durance, France*

(Received 21 August 2009; revised manuscript received 23 November 2009; published 4 January 2010)

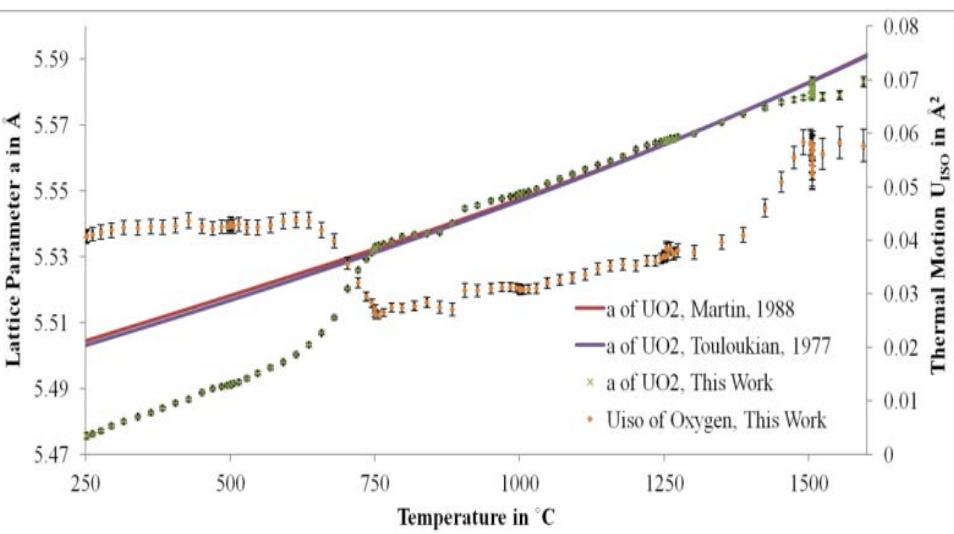
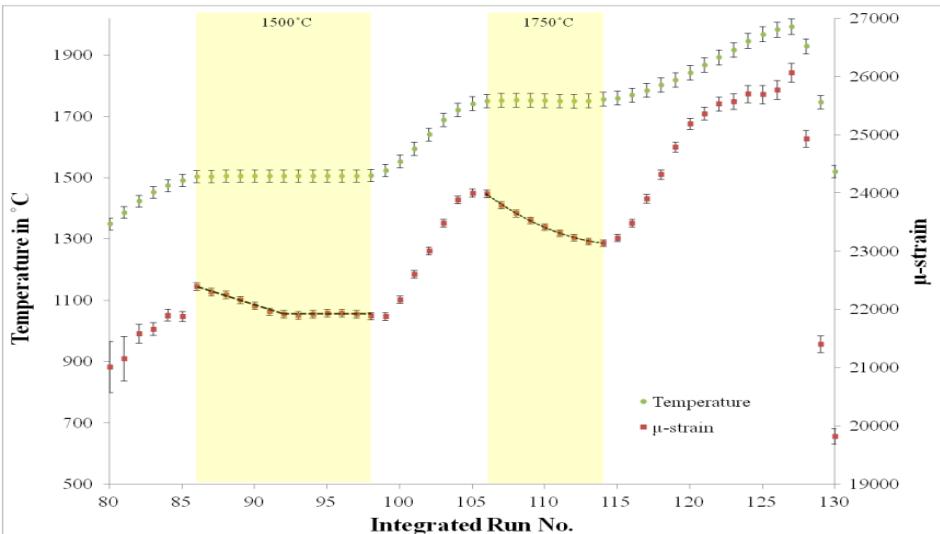
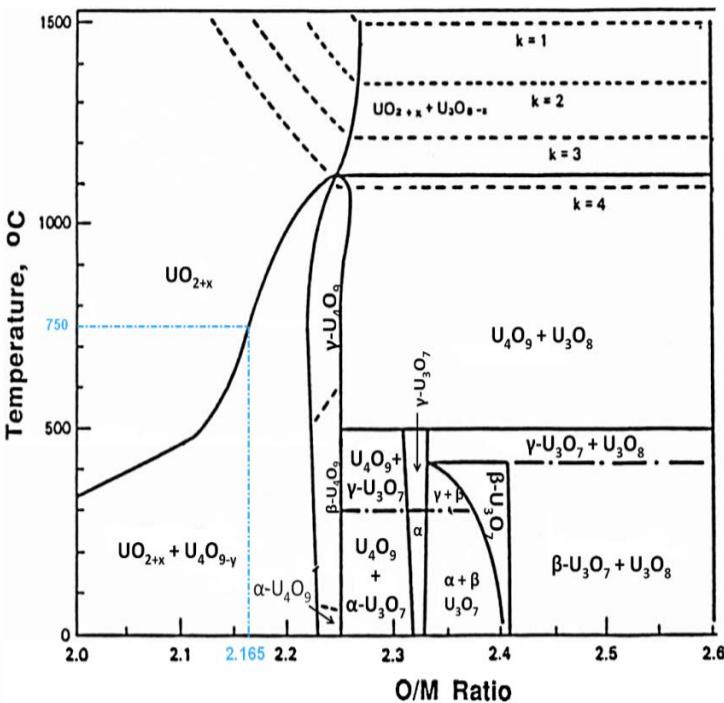
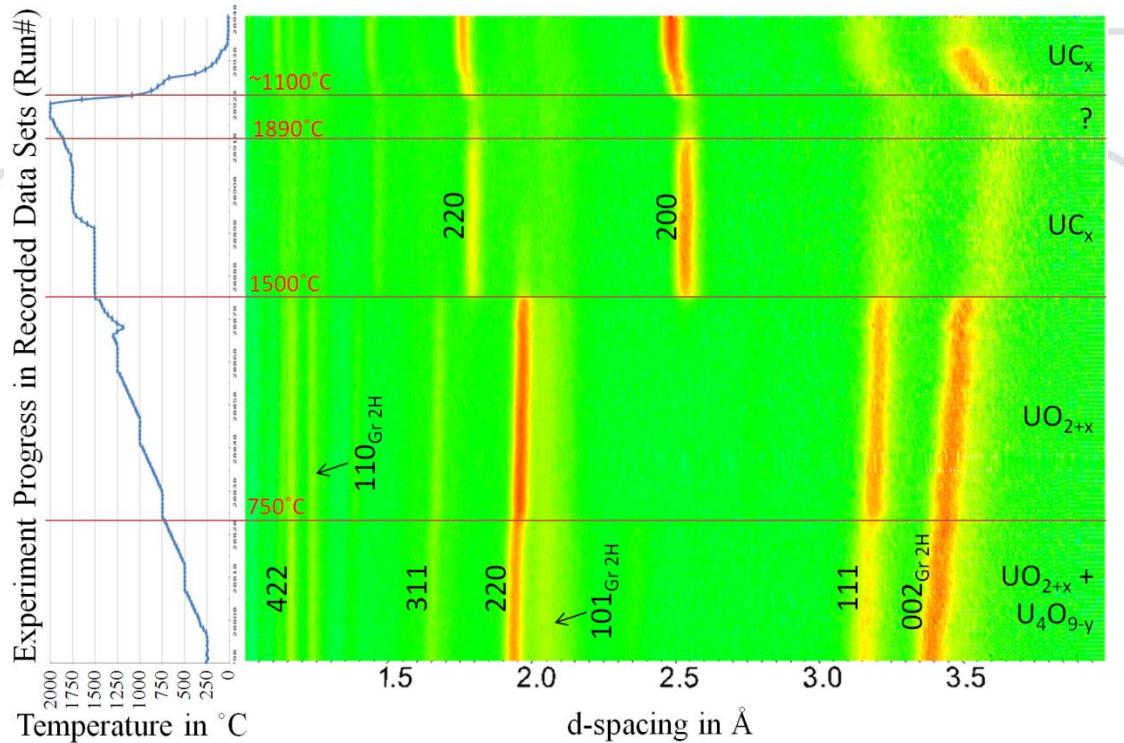
# Technique: Neutron Diffraction

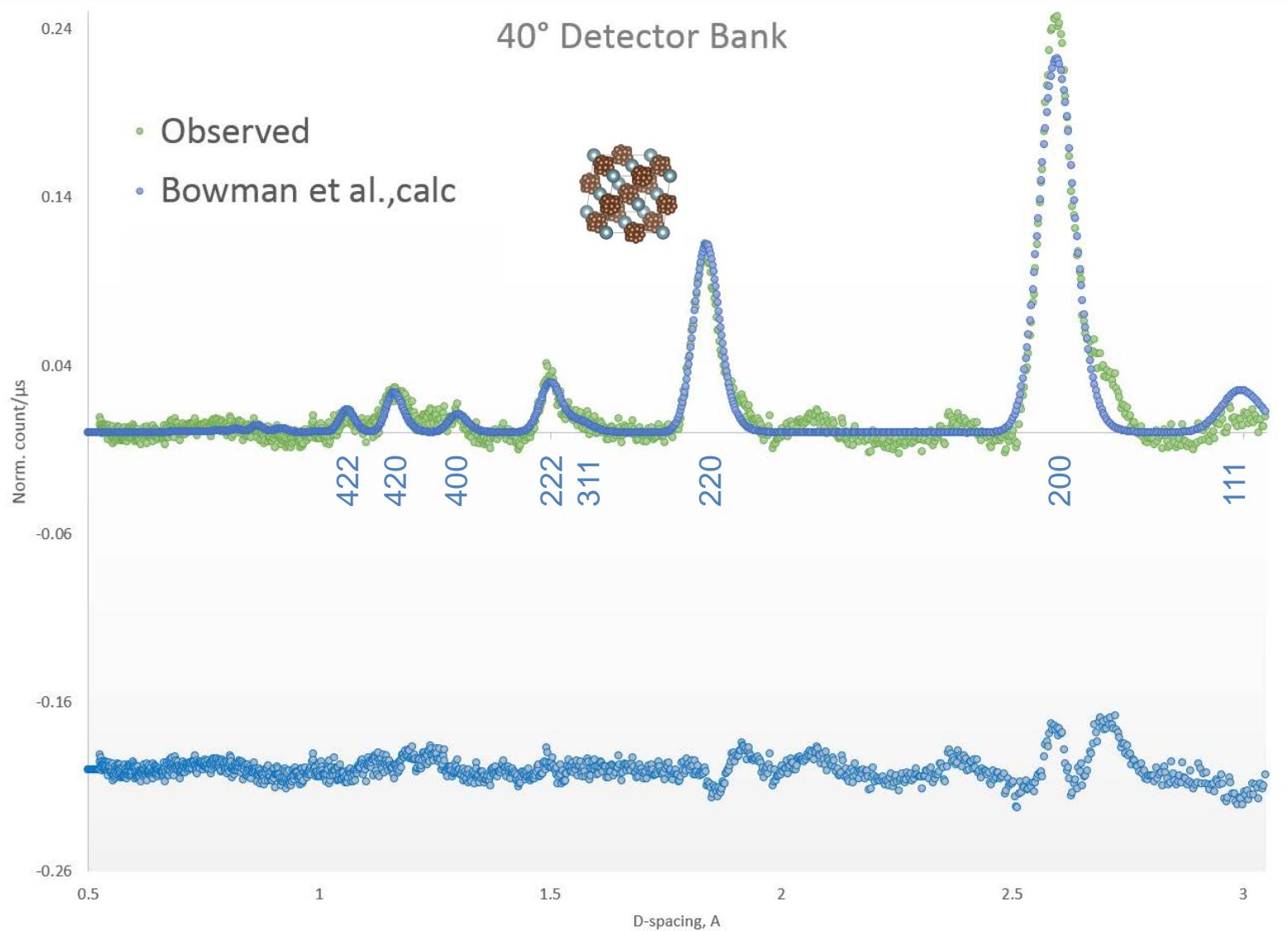


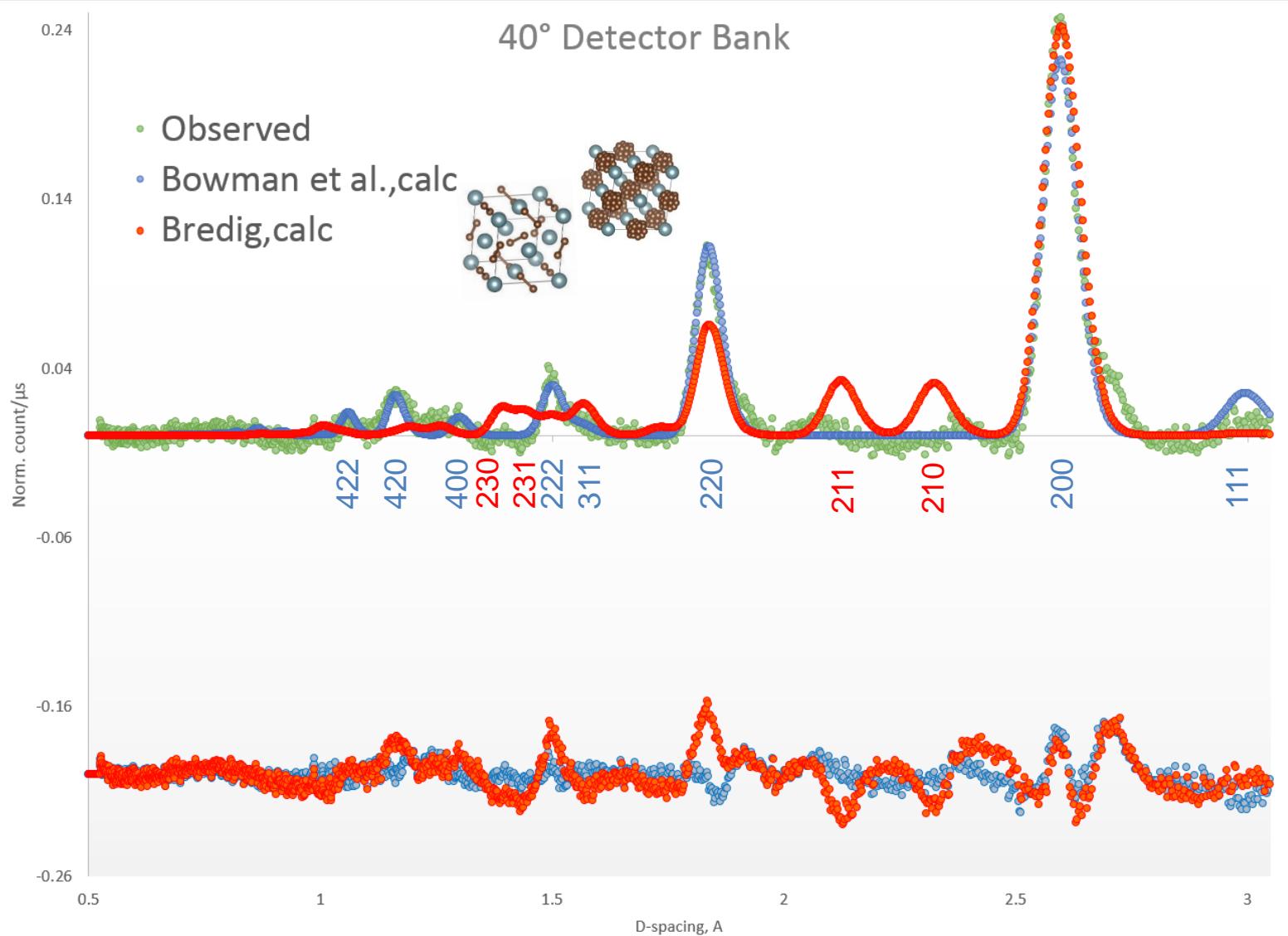
- HIPPO: High Pressure – Preferred Orientation
- Thermal flux:  $\sim 10^7 \frac{n}{s \text{ cm}^2}$  due to 8.83 m to target
- 51 detector panels with 1200  $^3\text{He}$  tubes, 4.9 m<sup>2</sup>
- Custom furnace:  $T_{\text{Max}} \sim 2500 \text{ K}$ ,  $\Delta T=100 \text{ K/min}$
- Pulsed Neutron Source / TOF
  - ➔ Same Bragg peaks occurs in each detector bank.  
Peaks present in only one bank indicate contamination.
- Large detector coverage allows to follow fast reaction kinetics
  - ➔ In situ characterization of
    - reaction kinetics and temperatures
    - crystal structure

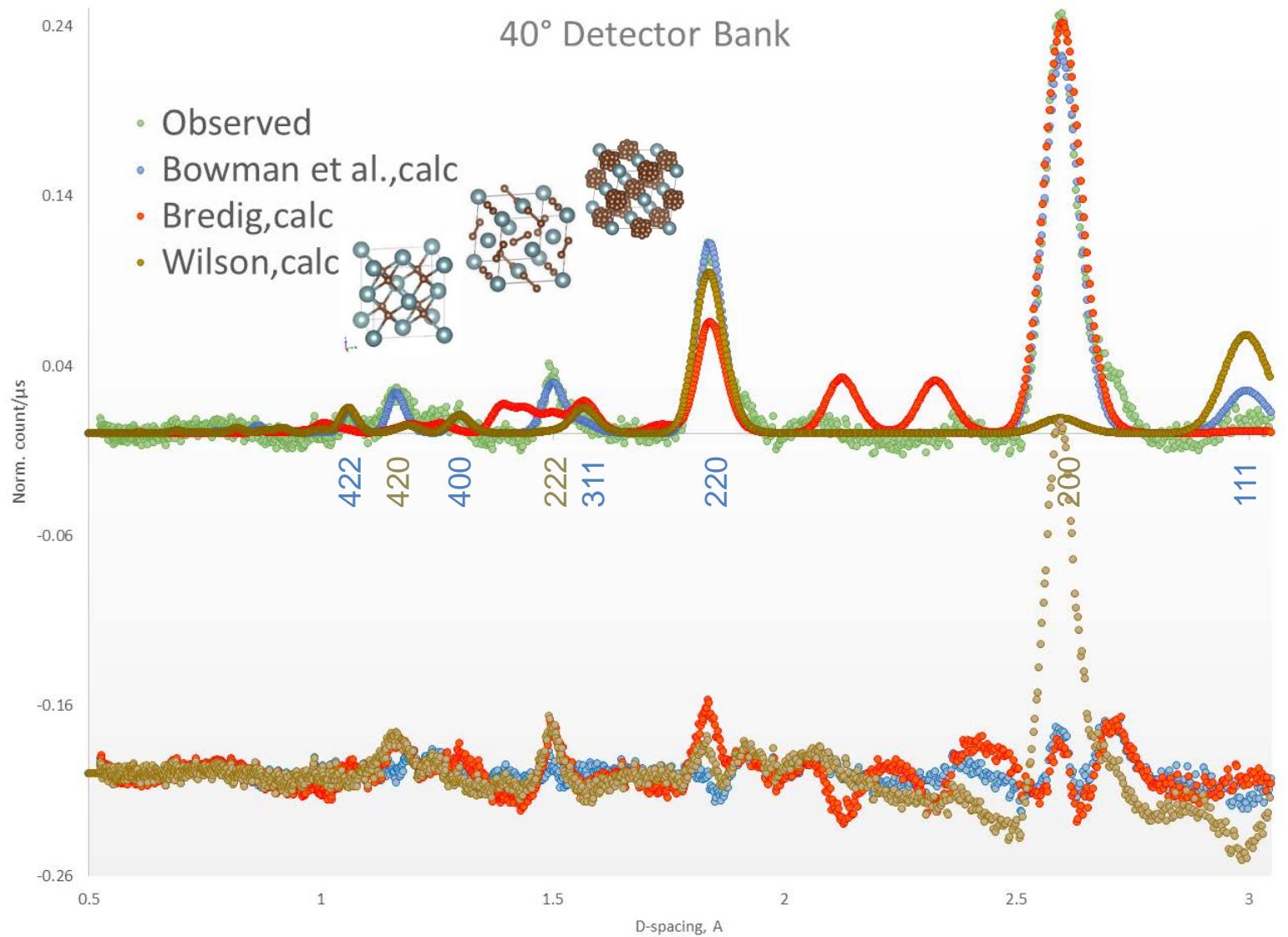


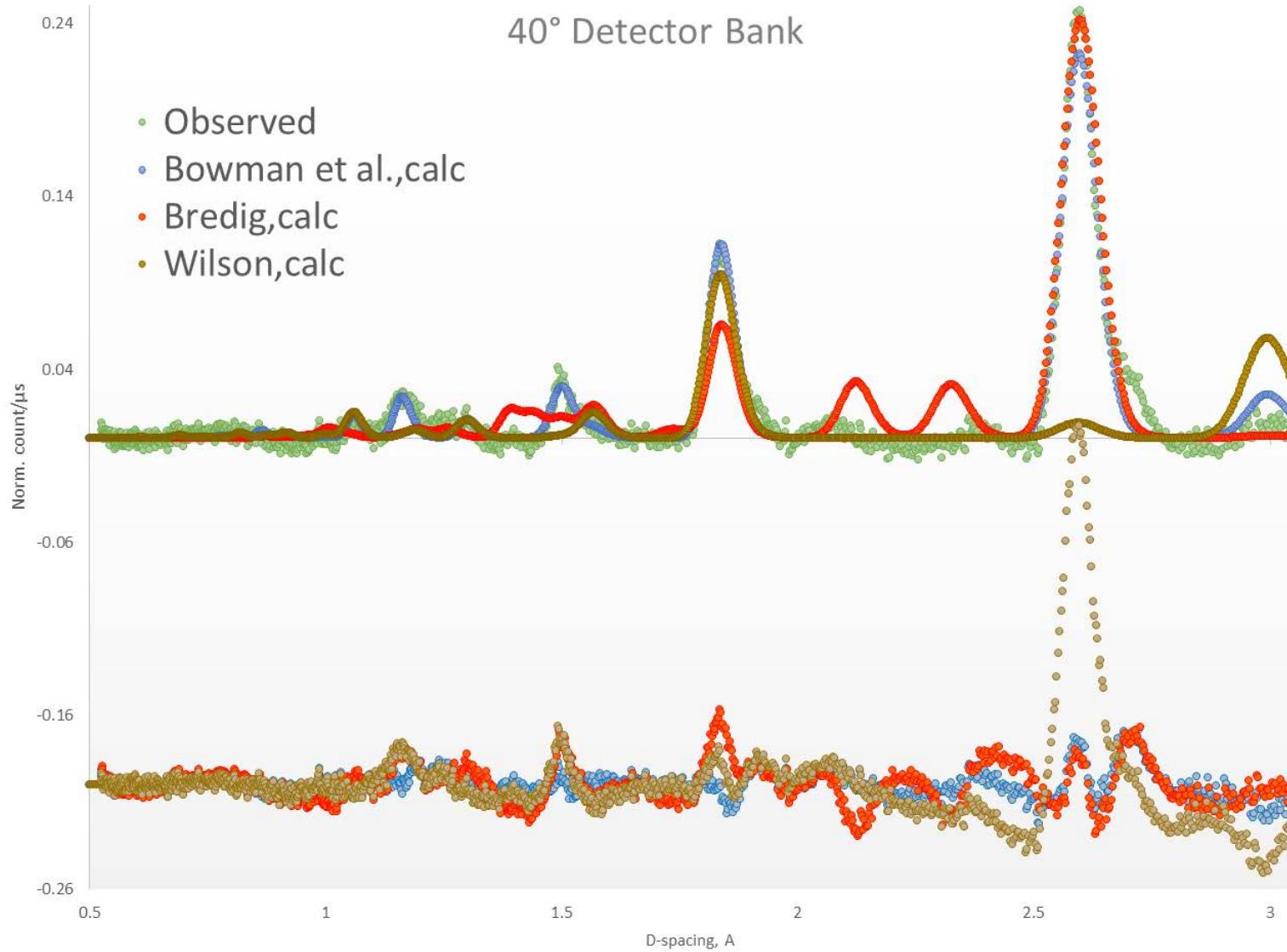
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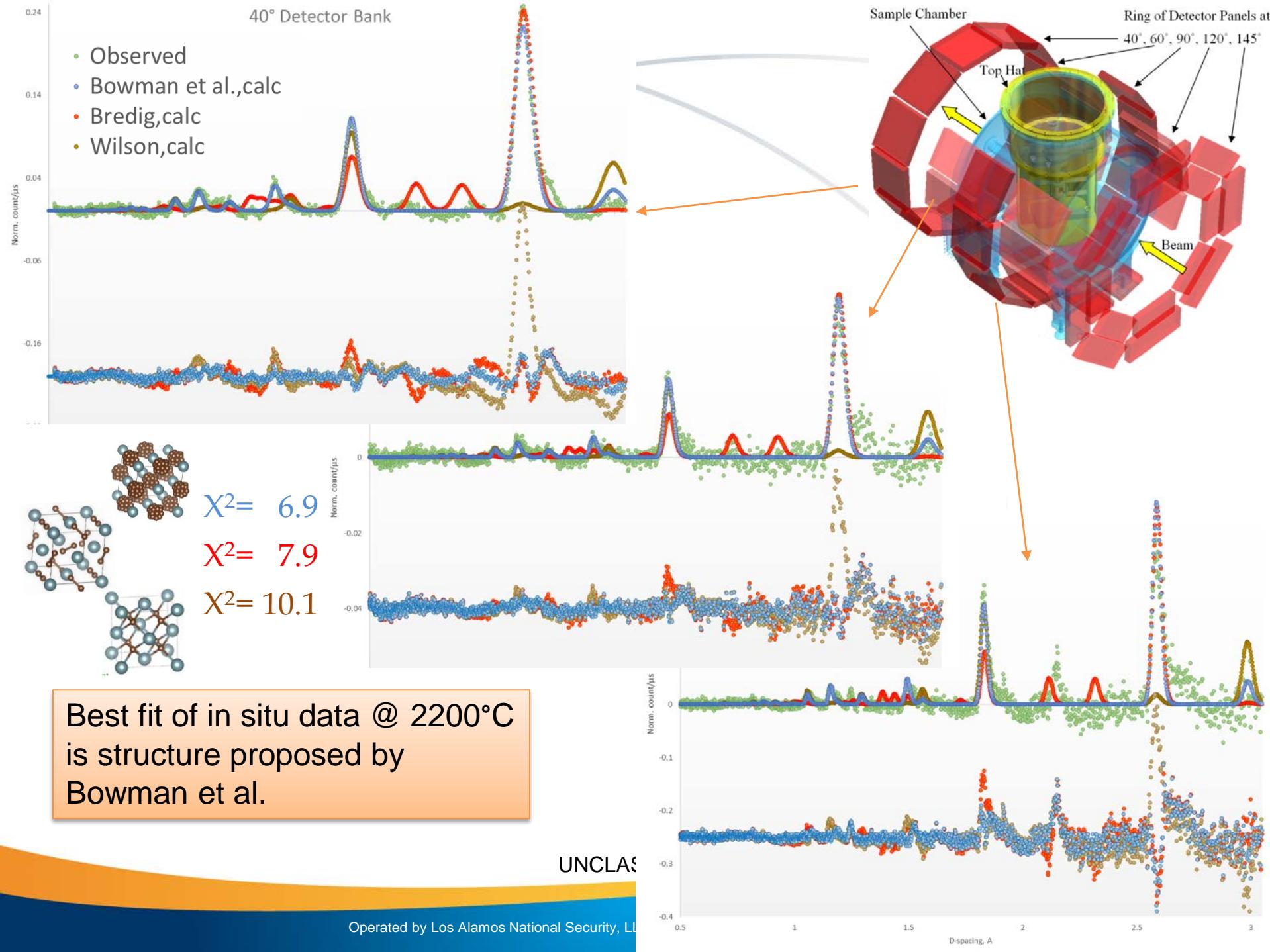




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## 40° Detector Bank

- Observed
- Bowman et al., calc
- Bredig, calc
- Wilson, calc



# Summary



- U-C system is actively researched, however, we present first *in situ* data since 1960. We identified the none quenchable, cubic,  $\delta$ -phase which in turn is fundamental to computational methods
- Rich datasets of the formation synthesis of uraniumcarbide yield kinetics data which allow to benchmark modeling, thermodynamic parameters etc.
- Order-disorder transition (carbon sublattice melting) observed due equal sensitivity of neutrons to both elements. This dynamic has not been accurately described in some recent simulation-based publications, good agreement with theoretical work by X.D Wen et al.
- LA env **Inorganic Chemistry** Article [pubs.acs.org/IC](https://pubs.acs.org/IC) sample

## Rotational Rehybridization and the High Temperature Phase of $UC_2$

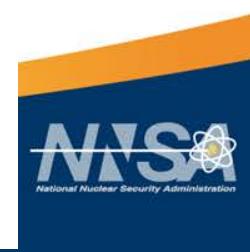
Xiao-Dong Wen,<sup>†</sup> Sven P. Rudin,<sup>†</sup> Enrique R. Batista,<sup>†</sup> David L. Clark,<sup>‡</sup> Gustavo E. Scuseria,<sup>§,||</sup> and Richard L. Martin<sup>\*,†</sup>

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<sup>||</sup>Chemistry Department, Faculty of Science, King Abdulaziz University, Jeddah 21589, Saudi Arabia



# Thank you

- My supportive Family (pics)
- Dr. Sven Vogel, Instrument Scientist/Data Analysis
- Dr. Luke Daemon, Sample Synthesis
- Eric Larson, CAD Design



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# High-Temperature Furnace: Motivation

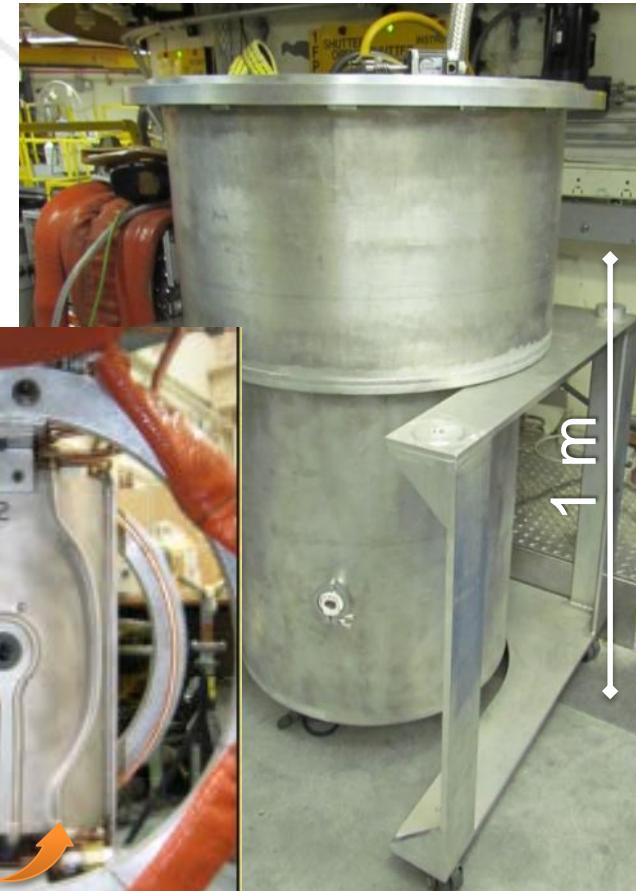
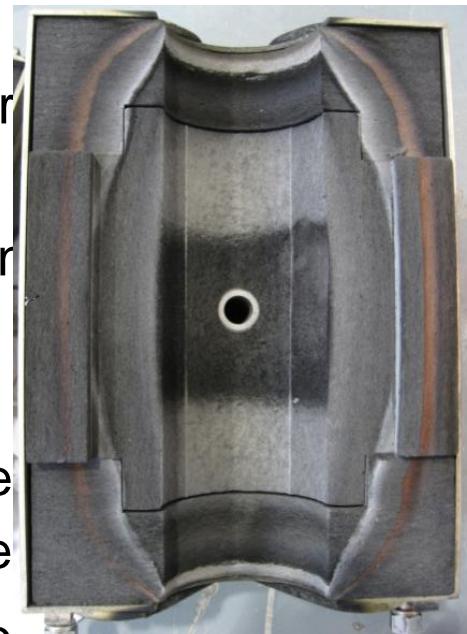
- Nuclear fuels are generally actinides (Th, U, Pu) with a light element (O,C,...) with  $T_m > 2500^\circ\text{C} \rightarrow \cancel{\text{XRD, Synchrotron}}$
- Phase transition kinetics require fast acquisition time
- Some models predict a  $\text{UO}_2$  fuel centerline of  $T > 1850^\circ\text{C}$  for SCWR
- Many materials have non-quenchable high temperature phases that are poorly known (e.g. cubic  $\text{UC}_2$ )

Facility	Beamline	High Temperature	
ILL, France	D20	ILL: 1150°C (1500°C Nb setup)	radio source allows for
LANSCE, NM	HIPPO	ILL: 1000°C	
HFIR	HB-2A	ILL: 800°C (1340°C Nb setup)	of $T > 1500^\circ\text{C}$
SNS, TN, USA	---	<under development>	
ISIS, UK	GEM	1100°C	
J-PARC, Japan	iMATERIA	<under development>	

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# High Temperature Furnace: Design

- $T_{Max} \sim 2500$  K,  $\Delta T=100$  K/min
- 360° sample rotation
- 50 mm sample height adjustment
- Heat shield and element made of graphite
- SCR controlled transformer  
10V@2000A
- Resilient safety interlocks
  - Water flow
  - Vacuum
  - Temperature
  - LabVIEW heating
- SCADA via LabVIEW, PID, control handling and notification



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	Space Group	Structure Type	Pearson Symbol	Cell Size	Observation Temperature
Bowman et al.	$Fm\bar{3}m$	Defect NaCl*	cF8	5.488 Å	1900°C
Wilson, W.B. (Freyss et al.)	$Fm\bar{3}m$	$CaF_2$	cF12	5.410 Å	1820°C
Bredig, M.A. (Nickel and Saeger)	$Pa\bar{3}$	$FeS_2$	cP12	5.472 Å	1820°C

\*Although listed as U0.5C in the ICSD, Bowman actually proposed UC2 with C2 molecules on the C sites of the NaCl structure, following the motion of a free rotator model or a random disorder model (oriented along [111] direction).

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Computational Materials Science 40 (2007) 562–568

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[www.elsevier.com/locate/commatsci](http://www.elsevier.com/locate/commatsci)

## Classical molecular dynamics simulation of uranium monocarbide (UC)

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Received 19 January 2007; received in revised form 19 February 2007; accepted 21 February 2007

Available online 16 April 2007

Uranium monocarbide (UC) is an important ingredient in the Indian nuclear fuel program. In Fast Breeder Test Reactor (FBTR), Kalpakkam, India; UC–PuC mixed carbide fuel has been used extensively for the last 15 years. Initially the FBTR was made critical with Mark I fuel (30%UC + 70%PuC) and later on it was continuously fuelled by Mark II (45%UC + 55%PuC) with the expansion of the FBTR core. Clearly, thermophysical and thermomechanical properties of UC are of special interests to us for better understanding and prediction of in-reactor fuel performance. Moreover, fuel behaviour in some sup-

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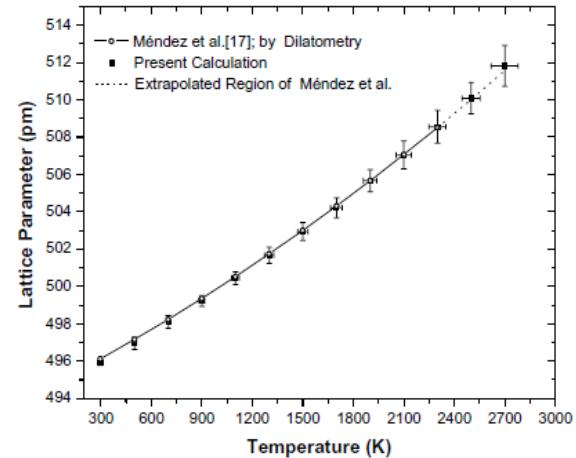


Fig. 2. Variation of lattice parameter of UC with temperature.

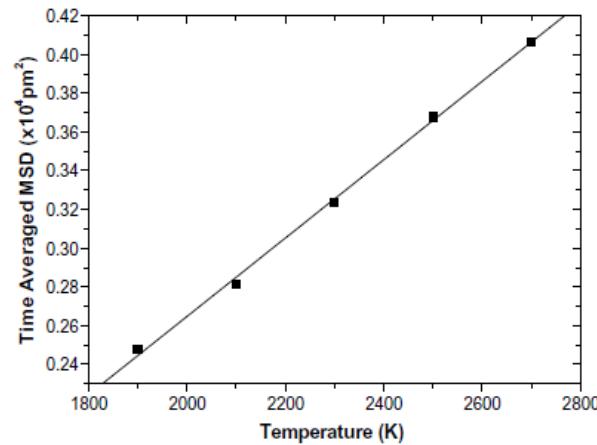


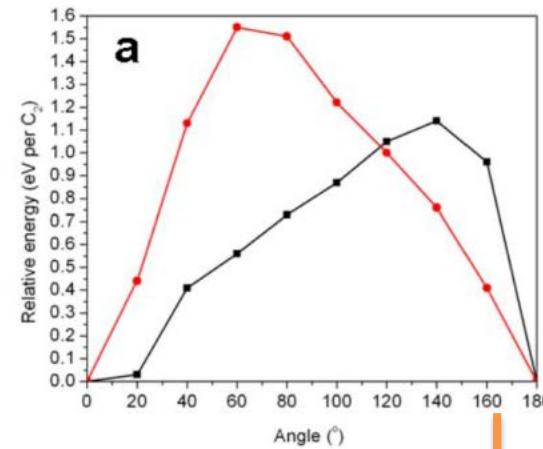
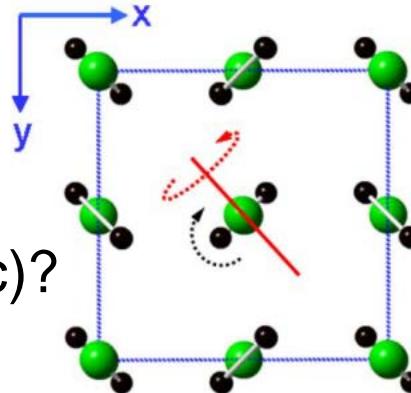
Fig. 6. Variation in time-averaged MSD of carbon ion in UC with temperature.

# Order-Disorder Transition

Why loss of intensity?

1. Melt?
2. Bredig Transition (Superionic)?
3. Motion of  $\text{C}_2$  Molecule
  - A. 'Free Rotation Model' by Bowman et al., 1966, ND.
  - B. 'Oscillating' Model by Wen et al., 2012, simulation.

→ Collaboration with Wen et al. to calibrate their HSE model with our experimental data



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