

FINAL REPORT

**Investigation of Surface Treatments to
Improve the Friction and Wear of Titanium
Alloys for Diesel Engine Components**

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prepared for the

**U.S. Department of Energy
Office of Energy Efficiency and Renewable Energy
Propulsion Materials Program**

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Foreword

This final report summarizes experimental and analytical work performed under an agreement between the U.S. Department of Energy, Office of Energy Efficiency and Renewable Energy, Office of Transportation Technologies, and UT-Battelle LLC. The project was directed by Jerry Gibbs, of the U.S. Department of Energy's Propulsion Materials Program, with management by D. P. Stinton and J. A. Haynes of ORNL. Participants included Peter J. Blau (Principal Investigator), Kevin M. Cooley (senior technician), Melanie J. Kirkham (materials scientist) of the Materials Science and Technology Division of ORNL, and Dinesh G. Bansal, a post doctoral fellow employed by Oak Ridge Associated Universities (ORAU) and who, at the time of this writing, is an engineer with Cummins, Inc.

This report covers a three-year effort that involved two stages. In the first stage, and after a review of the literature and discussions with surface treatment experts, a series of candidate alloys and surface treatments for titanium alloy (Ti-6Al-4V) was selected for initial screening. After pre-screening using an ASTM standard test method, the more promising surface treatments were tested in Phase 2 using a variable loading apparatus that was designed and built to simulate the changing load patterns in a typical connecting rod bearing. Information on load profiles from the literature was supplemented with the help of T.C. Chen and Howard Savage of Cummins, Inc. Considering the dynamic and evolving nature of materials technology, this report presents a snapshot of commercial and experimental bearing surface technologies for titanium alloys that were available during the period of this work. Undoubtedly, further improvements in surface engineering methods for titanium will evolve.

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No proprietary information has been included in this report.

Peter J. Blau
Oak Ridge National Laboratory
September 2012

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1.0 Executive Summary

The fuel efficiency of ground vehicles, like heavy trucks, can be improved by reducing their weight, minimizing internal friction losses, improving aerodynamics, and lowering rolling resistance with advanced tires. The current effort addresses lowering both engine component weight and friction losses. While primarily developed for aerospace structures, titanium alloy Ti-6Al-4V has the potential to replace heavier steel and cast iron in friction and wear-critical diesel engine components like connecting rods, movable turbocharger vanes, pistons, and even crankshafts. While Ti-6Al-4V exhibits excellent corrosion resistance, good fatigue strength, and satisfactory fracture toughness, it has poor friction and wear characteristics. Thanks to the aerospace industry's interest in reducing fretting wear in turbine engines, a variety of approaches are available to improve the friction and wear of Ti contact surfaces. Unfortunately, the majority of these are impractical for diesel engine connecting rod bearings because such surfaces must withstand well over a billion lifetime sliding contact cycles and be compatible with conventional lubricants. Using conventional inserts like plated bronze for big end bearings in Ti alloy connecting rods is a lower-risk, near-term solution to light-weighting engines, but some bronzes contain lead, a material with negative environmental impact. The current project investigated whether a Ti alloy could be surface engineered to avoid the need for inserts, and in addition, extend its possible use to other bearing-critical surfaces in the engine like cam lobes which also experience moving contact.

A two phase effort was conducted. During Phase 1, more than twenty surface engineering processes were compared in engine-conditioned diesel oil in a reciprocating pin-on-flat (RPOF) apparatus (ASTM Standard G133). Candidate treatments included diffusion treatments (carburizing, nitriding, oxidizing), the application of hard coatings (TiN, CrN, diamond film), application of a plated coating used on actual bearing inserts (Cu-Ni-In), a Ti-matrix TiB₂ in situ-formed composite, and mechanical working (by shot peening or hammer planishing). In addition to studying the effects of individual surface treatments, some treatments were combined in an attempt to discover possible synergistic benefits. X-ray studies indicated that subsurface microstrain levels varied depending on the combination of treatments. Of the initial set tested in Phase 1, five promising treatments were down-selected for further investigation in Phase 2.

A variable-loading bearing tester (VLBT) was designed and built. Programmed to simulate the time-varying load spectrum of a typical connecting rod, the VLBT was used during Phase 2 to compare the frictional behavior and durability of promising surface engineering treatments under the range of lubrication regimes that are typical of engine connecting rod bearings. The treatments down-selected in Phase 1 did not produce the same rankings when they were tested using the VLBT because it used a different contact geometry, unidirectional motion, and higher speed than the Phase 1 procedure. VLBT data were more typical of the mixed or hydrodynamic lubrication regimes than were the RPOF data which typified boundary lubrication conditions. Implications for these results for applications like connecting rod bearings are discussed. The RPOF test of Phase 1 stressed the materials more than did the VLBT test of Phase 2 which operated under nearly full-film lubrication. In fact, some treatments that failed to pass Phase 1 criteria, but were added to the test matrix for comparison, ranked higher under VLBT conditions. Clearly, test methodology can emphasize either material characteristics or lubrication characteristics. As a result, different tests produce different friction and wear rankings for the same set of sliding partners.

Results of Phase 1 were published in the proceedings of the 18th International Conference on Wear of Materials, and results from Phase 2 have been accepted for the 19th International Conference on Wear of Materials (2 separate papers). These data support a series of recommendations for diesel engine designers interested in light alloys.

A key finding from the current work is that most of the coatings and surface treatments employed here affect only a few thousandths of an inch (small fractions of a millimeter) deep in the bearing surface, and once this thin zone is worn through, the exposed Ti substrate exhibits poor friction and wear behavior. In addition, such treatments tend not to be self-healing or easily restorable; therefore, there is little alternative but to replace the entire part (or machine it out and add inserts). Based on current results, two surface engineering treatments offered the best combination of resistance to material contact effects (Phase 1 tests) and performance under well-lubricated conditions (Phase 2 tests). The first was the commercial chromium nitride (CrN) coating known as FortiPhy™, and the second was an experimental high velocity powder consolidated (HVPC) coating from Pennsylvania State University containing Ni with a B additive. The latter choice was based on its ranking during lubricated friction and wear results.

In summary, one conservative approach to the use of Ti in friction and wear-critical engine components is simply to avoid the use of bare or treated Ti surfaces entirely, and use inserts of traditional bearing materials. The case in point is for lightweight connecting rods, but in other cases, the results of the current work suggest a potential for the use of surface treated or thin layer coated titanium alloys in components that operate under mixed film or thick film lubrication conditions where resistance to solid/solid moving contact is less of an issue.

2.0 Statement of Objectives

The overall goal of this effort is to facilitate a 50% increase in heavy vehicle freight efficiency (ton-miles per gallon). To support this goal, there is an effort to increase the use of lightweight structural materials, like Ti-based alloys, in certain engine components which require low friction while also resisting wear. Since bare Ti is unacceptable for bearing surfaces, the solution will require surface engineering treatments or coatings. The current effort focuses on identifying the most promising ones, be they experimental or commercialized.

The objective of this effort is to provide guidance to diesel engine designers for how titanium-based alloys could, by virtue of advanced surface engineering practices or coatings, be used for engine components that require low friction surfaces and high wear resistance. The large end bearing on a connecting rod is a particular focus of this work, but other potential applications include exhaust gas recirculation valves, tappets, valves and valve guides, crankshafts, and piston skirts. The ultimate purpose is to anticipate future needs for lightweight materials like titanium so that the diesel industry's introduction of new technology can be accelerated.

3.0 Benefits to the DOE Office of Transportation Technologies Mission

The introduction of advanced materials into transportation has historically taken from 5-20 years, depending on the maturity of the technology, the urgency of the need for property improvement or cost control, the ability or inability of current materials to meet design requirements for performance and durability. The greater the engineering drivers, the shorter the material introduction cycle, especially if current materials do not suffice. A great deal of federal investment has been made in lightweight materials technologies for transportation, material classes have included metals, ceramics, and composites based on a wide variety of matrix materials. Some of these materials have been commercialized and others with encouraging performance characteristics still wait in the wings, pending cost reduction. Relative to steels and cast irons, the density of metals like titanium, aluminum, and magnesium offer

potential weight savings if properly employed in engine and drivetrain components. For example, ORNL has previously published assessments of the use of titanium alloys for heavy truck brakes [1,2]. As with any commercial material technology, there must be a trade-off between performance and cost. Sometimes, the need to meet industry regulatory requirements overrides cost penalties.

4.0 Technical Introduction, Approach, and Results of Experimental Work

4.1 Introduction and Approach

While Ti alloys have been successfully employed in a wide variety of structural applications – from aircraft* and submarine components to high-end sports equipment and biomedical implants – Ti alloys have not been widely used in ground transportation components. Among the main reasons for its relative scarcity in automobiles and trucks has been material cost and availability in an industry in which a few pennies per vehicle can influence material selection. The cost penalty for Ti could be off-set if (a) raw material and manufacturing costs were decreased, and (b) the performance and fuel efficiency benefits of Ti alloys' light weight, specific stiffness, and corrosion resistance off-set its cost.

Like aerospace components and sports equipment, heavy vehicle engines and drive trains could be light-weighted using Ti, but there is an additional problem to be considered. Namely, Ti alloys, like other light alloys based on Al and Mg, suffer from a lack of good friction and wear characteristics and that make use in bearings, gears, and other kinds of moving surface contacts problematic. However, the technical community has not quite given up on Ti for bearing surfaces and in fact has invested considerable effort in surface engineering Ti alloys using coatings and various kinds of diffusion treatments. Much of this work has been focused on aerospace applications – notably fretting of Ti in aircraft turbine engine blades and discs. Relatively little work has been aimed at other kinds of wear parts and types of wear in lubricated situations. Specific surface-critical applications include piston/cylinder contact surfaces, roller followers, crank shafts, valve guides, valve seats, and connecting rods. The latter application is the focus of the current work, whose intent is to identify promising approaches to improve the surface durability and lubricity of Ti alloys.

The parts and features of a typical diesel engine connecting rod are shown in Figure 1. A diesel engine connecting rod from a commercial diesel engine is also shown in Figure 2. There are two primary bearings. The upper, small end bearing experiences reciprocating sliding between the bearing and wrist pin as the piston oscillates, while the big-end bearing experiences continual motion in one direction.

Bearing Materials. The design of most connecting rod bearings involves the use of inserts. In order to provide a combination of stiffness, running-in performance and lubricity, such inserts are often composed of a number of layers and platings, depending on the engine manufacturer [3]. Copper-based alloys (bronzes), like those listed in Table 1, have historically been used connecting rod bearings [4]. Note that many of them contain lead (Pb), making recycling or wear debris for such materials a concern for environmental reasons. It would be desirable to use Ti alloy other than a leaded bronze for that reason, if such bearings could be made to work.

(* Each Boeing 787 aircraft contains as much as 100 tons of Ti-alloys. See: www.commoditymarketing.com.au/titanium.php.)

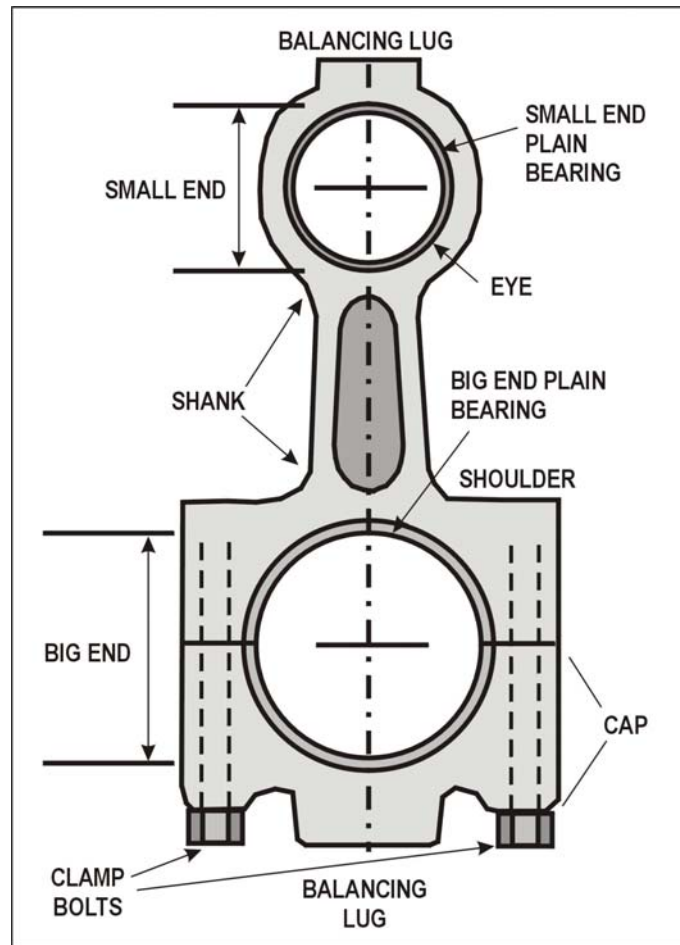


Figure 1. Parts of a typical connecting rod.



Figure 2. Commercial diesel engine connecting rod.

Table 1.
Bearing and Bushing Alloys used for Connecting Rods
(Adapted from ref [4], Table 11-11.)

Category	Designation	Composition
Aluminum-based	SAE 770	6.25 Sn, 1 Cu, 1, Ni, bal. Al
	SAE 780	6 Sn, 1, Cu, 1.5 Si, 0.5 Ni, bal. Al.
	SAE 781	4 Si, 1 Cd, bal. Al
Copper-based	SAE 48	70 Cu, 30 Pb
	SAE 49	76 Cu, 24 Pb
	SAE 480	65 Cu, 35 Pb
	SAE 481	60 Cu, 40 Pb
Tin-based	SAE 11*	87.5 Sn, 6.75 Sb, 5.75 Cu
	SAE 12*	89 Sn, 7.5 Sb, 3.5 Cu
Lead-based	SAE 13*	6 Sn, 10, Sb, bal. Pb
	SAE 14*	10 Sn, 15 Sb, bal. Pb
	SAE 15*	15 Sb, 1 Sn, 1 As, bal. Pb

*Note: ‘Babbitt’ alloy. Babbitt alloys comprise more than one composition, and are intended to serve as bearing materials. They are mixtures of Cu, Sn, Pb, and Sb. They were named eponymously after Isaac Babbitt who invented them in the early 1800s.

A cross section of a commercially insert for a big end bearing is shown in Figure 3. The top layer is a thin plated coating of about ~ 11-12 μm in thickness. Below it is a bronze layer of thickness ~ 248-250 μm , and below that (blue in the image) is the steel backing plate. The light and dark gray phases in the interdendritic areas of the bronze are Pb that is added for lubricity.

Knoop microindentation hardness data for of the various layers shown in Figure 3 are given in Table 2. Note that there is a factor of about 10 in hardness difference between the soft upper plating and the backing plate, with a bronze layer having intermediate hardness. The soft plating tends to wear off during the running-in period and expose the underlayer for the rest of the service lifetime. Running-in, when done properly establishes the surface conditions for successful long-term operation.

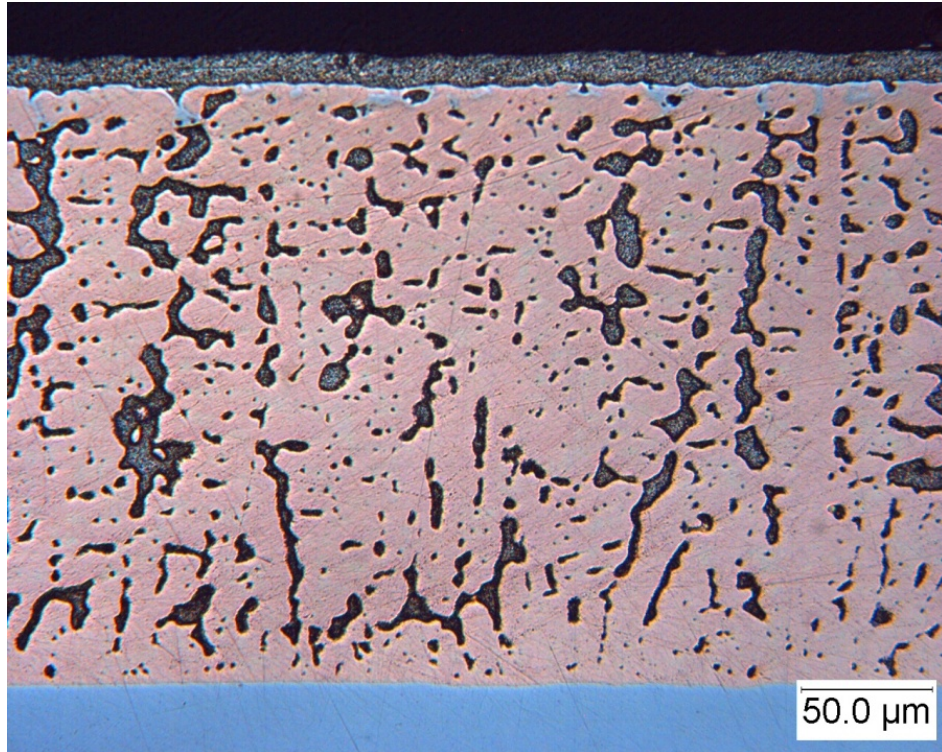


Figure 3. Polished cross-section of a large end bearing insert (polarized light).

Table 2. Knoop Microindentation Hardness of Layers in a Bearing Insert

Material	Test Load (g-f)	HK, average (kg/mm ²)	HK, average (GPa)
Steel backing	10	215.1	2.11
	25	230.4	2.26
	100	219.7	2.16
Bronze, matrix phase	10	107.9	1.06
	25	114.1	1.12
Top coating layer	10	28.2	0.28

Crankshaft Materials. The friction and wear of a plain bearing or bushing depends upon the successful mating of opposing materials. In the current case, the engine crankshaft rotates inside each connecting rod bearing. Several kinds of forged and heat-treated steels are commonly used (see Table 3). The choice of the crankshaft material is dictated by engine-specific characteristics, cost, availability, machinability, and other engineering factors.

Table 3.
Typical Crank Shaft Alloys
(Compositions given in weight %, Source: [5])

Alloy Type	Composition / Description	Approximate Hardness
Mn-Mo steel	Composition: 0.38 C, 1.5 Mn, 0.3 Mo. Forged, oil quenched and tempered. Compatible with bearings plated with Al-Sn or Cu-Pb	250 Brinell
1%-Cr-Mo Steel	Composition: 0.4 C, 1.2 Cr, 0.3 Mo. Forged, oil quenched and tempered. Can be flame or induction hardened in addition. For highest hardness, can be nitrided. Compatible with bearings plated with Al-Sn or bronze.	280-480 Brinell
2.5%-Ni-Cr-Mo Steel	Composition: 0.31C, 2.5 Ni, 0.65 Cr, 0.55 Mo. Forged, oil quenched and tempered. Favored in heavy duty diesels.	300 Brinell
3%-Cr-Mo or 1.5%-Cr-Al-Mo Steel	Compositions: (a) 0.15 C, 3 Cr, 0.5 Mo; (b) 0.3 C, 1.5 Cr, 1.1 Al, 0.2 Mo. Oil quenched and tempered. The shafts are also case-hardened by nitriding. Favored in heavy duty diesels.	(a) 800-900 HV (b) 1050-1100 HV
Nodular cast irons	Composition: 3 – 4 C, 1.8 - 2.8 Si, with graphite nodules dispersed in a pearlite matrix. Ce or Mg are used to achieve this structure. Flame or induction hardening are used to increase surface hardness.	550-580 Brinell

In the present work, we chose a readily available alloy steel 8620 with an as-received hardness of about 223 Brinell. Its typical composition is as follows (wt%): 0.18-0.23 C, 0.4-0.6 Cr, 0.7-0.9 Mn, 0.15-0.25 Mo, 0.4-0.7 Ni, and was felt to be sufficiently close in composition to the steels in Table 2 to use as a counterface material.

Lubrication Regimes. In order to evaluate a series of candidate materials and surface treatment for application to a large end connecting rod bearing in a laboratory, it is important to understand the operating environment and introduce similar variables for screening. In the current case, we focused on relative motions in the contact and the variations in load that occur as such a bearing rotates. It was established by the work of Martens, Stribeck, and others (as described in [6,7]) that there exists a relationship between the normal force (P), lubricant viscosity (η), and relative velocity (v) of moving bearing surfaces given simple geometries of contact such as a bearing rotating in a journal. That work evolved into a powerful bearing design and analysis tool that is popularly known as the Stribeck curve (shown schematically in Figure 4). In bearing situations like connecting rods in which non-constant loads or speeds might occur, the operating point in the tribosystem in essence travels up or down the Stribeck curve, producing changes in the amount of surface contact, with more solid/solid contact being associated with higher wear and friction levels (up and to the left), and less contact being associated with the development of a thick lubricating film between opposing surfaces (at the bottom of the curve and to the right).

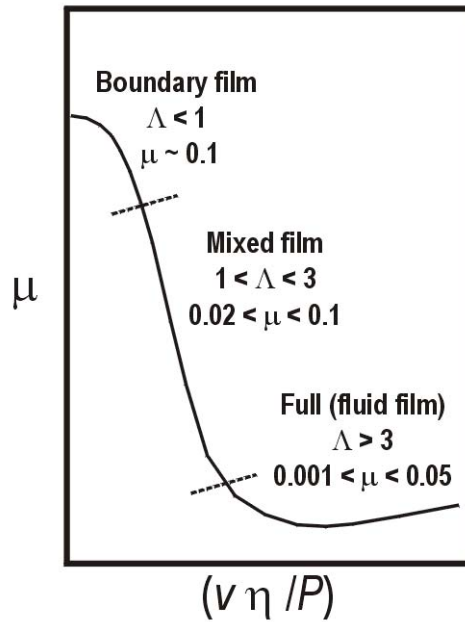


Figure 4. Portion of the Stribeck curve showing various regimes of friction and film thickness.

During the lifetime of a typical diesel engine, connecting rod bearing housings typically sustain as many as a billion or more stress cycles, ranging from the high compressive forces due to combustion and high tensile forces due to inertia [8].

The changing loads due to rotation of a connecting rod bearing in an automobile engine was reported by Taylor [9] in the form of a polar diagram in which bearing load is plotted as a function of the angle of rotation, with two full strokes equaling a single compression, combustion, and exhaust cycle. Such a curve was replotted in linear rather than polar formal in Figure 5. The relative changes in load with time served as a guide to programming the forces on the apparatus which is described later in this report.

The typical magnitude of the film thickness ratio (Λ), defined as the dimensionless ratio of lubricant film thickness to composite roughness, is also associated with certain lubrication regimes, as in Figure 4. It can be assumed that the farther up and to the left one travels on the curve, the more frequent and intense is the surface contact, with very little contact occurring in the mixed film region and more in the boundary film region. Thus, as operating conditions for a transient tribo-contact like an engine bearing or piston on liner change during the course of operation, the degree of solid contact and the probability of wear at any instant of time also vary.

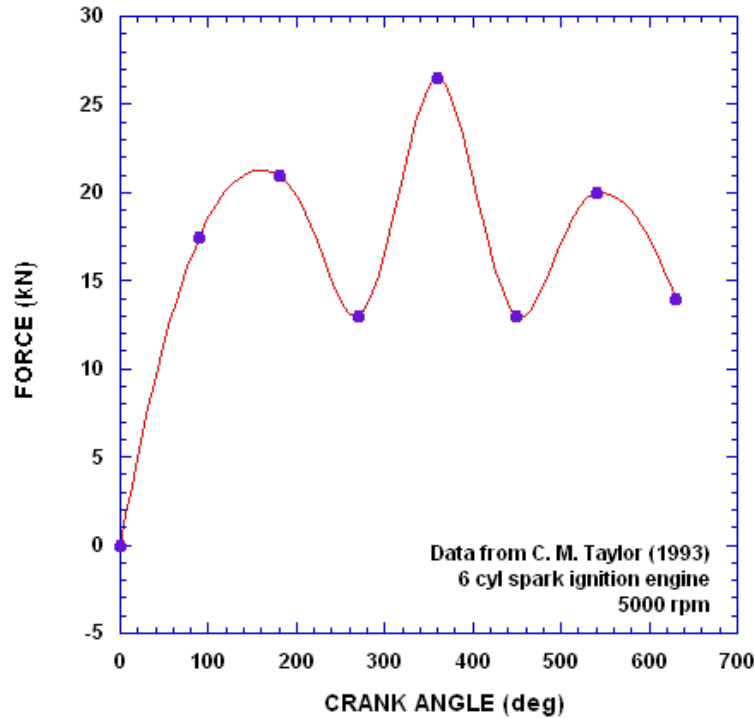


Figure 5. Bearing forces as a function of crank angle in a fired engine (based on data from [9]).

Research plan. This project was comprised of two phases. The first was a pre-screening and materials acquisition phase using existing friction and wear testing equipment, and the latter was intended as a better simulation of lubricated conditions. It used a specialized apparatus which was designed and built during the period of Phase 1.

Phase 1: Surface Technology Review and Pre-Screening Experiments. Following an in-depth literature review and discussions with industry contacts, over twenty surface treatments and coatings for Ti alloys were placed on a candidates list. Titanium alloys to be used as the substrate and a reference bearing material were obtained. ASTM standardized friction and wear tests (using ASTM G133, whose development was led by ORNL during the 1990s) were performed using engine-conditioned diesel drain oil. Also during Phase 1, a variable-loading ('spectrum loading') bearing materials test system (herein called the VLBT) was designed and built in order to provide the ability to simulate the variable contact and lubrication conditions experienced by connecting rod bearings in service. The ability to study such transient effects enabled more realistic screening of bearing performance than traditional friction and wear tests which operate under constant load and speed.

Phase 2: Simulative Testing in the Spectrum Loading Bearing Apparatus. The leading candidates for surface engineering Ti alloys, as determined from Phase 1, were tested in the VLBT under load spectrum conditions. A comparison could be made between friction and wear results from Phases 1 and 2 to determine the extent to which the choice of screen test protocol affected material rankings. The salient results, summarized here, provide the basis for a series of recommendations on using Ti alloys for bearing surfaces in engine components like connecting rod big end bearings, cam lobes, and others.

4.2 Phase 1 Test Results

A standard reciprocating pin-on-flat test was selected for the Phase 1 screening of the surface treatments because it provides high contact stresses and allowed comparison with previous studies under the same conditions. Such severe test conditions focused more on material effects than lubricant properties, and thus provided a more conservative approach in selection of surface treatments and coatings.

Phase 1 screening tests, whose details are also described in two papers arising from this work [10-11], were conducted using ASTM standard G133 [12], a linearly-reciprocating ball-on-flat geometry. Test parameters are given in Table 4. Procedure A was used for baseline materials of Ti64 and bearing bronze (CDA 932), while Procedure B was used for all the tests in a modified form by using only 3-4 drops of the engine conditioned 15W40 oil at room temperature. The lubricant used in these tests was drain oil from a standard Mack T-11 test [13] and contained soot and acids.

Table 4. ASTM G133 Linearly Reciprocating Pin-on-Flat Test Conditions

Variable	Procedure A	Procedure B
Applied load (N)	25	200
Stroke length (mm)	10	10
Oscillating frequency (cyc/s)	5	10
Test duration (s)	1000	2000
Sliding distance (pin) (m)	100	400
Atmosphere	Air	Air
Humidity range (%RH)	40-60	Not specified
Method of lubrication	None	Immersion
Temperature of lubricant (C)	Not applicable	150*
Air temperature (C)	19 – 25	Not specified
Slider ball diameter (mm)	9.53	9.53

* For this work, the lubricant was used at room temperature. Therefore, results are reported as being performed under a modified Procedure B.

Tests were conducted using a 1989 vintage commercial reciprocating test apparatus (Plint model TE77, Phoenix Tribology, UK). AISI 52100 high Cr bearing steel was used as the standard counterface (ball) material, but a few experiments were also conducted using a silicon nitride ceramic (NBD 200, Norton-St Gobain Ceramics). Friction measurements were made using data acquisition software (LabView™) at 200 Hz, recorded and averaged for 1 second at 5 second intervals. Therefore, there were 2×10^5 friction coefficient data points collected for each Procedure-A test and 4×10^5 points for each Procedure B (mod.) test. The surfaces of the flat specimens (25 x 25 mm square) were polished with 400-grit, 600-grit and 1 µm diamond paste. Both the ball and flat specimens were cleaned in acetone and isopropanol prior to testing.

Wear volumes for the flat test coupons were measured using a vertical scanning interferometer (VSI) (Wyko™ NT9100, Bruker Instruments) and processed using the accompanying software. Examples of images used for wear measurements are shown in Figures 6(a) and (b). The wear volume of the pin specimen (a fixed ball) was estimated by measuring the diameter of the wear scar on the tip under a microscope and then using the equations listed in ASTM G133.

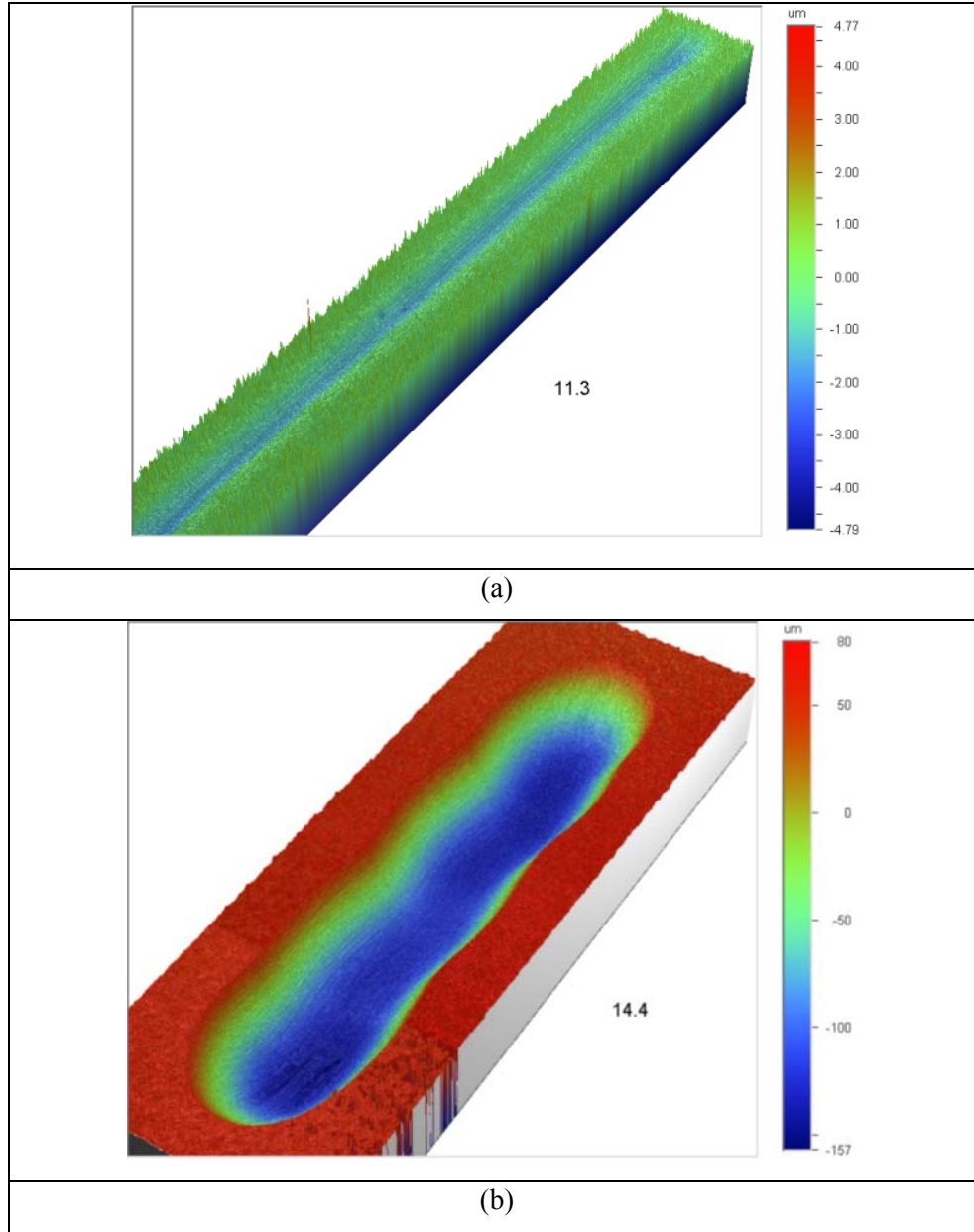


Figure 6. 3-D reconstructions of VSI images of wear tracks of (a) a completed test of OD Ti64, and (b) a prematurely terminated test of carburized and shot-peened Ti64 in which the wear was excessive.

Table 5 lists different materials and surface treatments investigated in Phase 1, along with corresponding Vickers and Knoop hardness (100 g-f load), the surface roughness parameters, and the identification codes of the test coupons used in Phase 1. Barring severe wear or premature test cessation, three tests were conducted for each candidate material/treatment. A description of each material, its treatment, and its source is provided in Appendix A. Indentation hardness values are the average of 8–12 indents per test coupon and the corresponding standard deviation is also listed. The arithmetic surface roughness (Ra) and root-mean square roughness (Rq) were measured at 6 – 8 locations on the flat test coupon surfaces using the same VSI that was used for wear measurements.

Table 5. Phase 1 Test Coupon Materials, Surface Finish, and Hardness

Category	Material or Treatment	Designation	Roughness R_a (μm)	Roughness R_q (μm)	HV (GPa)	HK (GPa)
Bulk materials	Ti-6Al-4V alloy	Ti64	0.067	0.087	3.64 ± 0.11	3.73 ± 0.11
	Bearing bronze	BB	0.032	0.05	1.1 ± 0.06	1.1 ± 0.06
	Nitinol (60Ni-40Ti)	60NiTi	0.011	0.015	3.81 ± 0.13	3.59 ± 0.15
Thermal and chemical treatments	Oxygen diffusion	OD	0.31	0.38	7.1 ± 0.76	8.59 ± 0.5
	Nitriding	NR	0.12	0.157	6.96 ± 0.26	7.37 ± 0.15
	Carburizing	CR	0.124	0.16	6.92 ± 0.31	7.03 ± 0.22
	Nitinol + heat treatment	60NiTi + HT	0.3	0.376	10.11 ± 0.73	15.55 ± 0.75
	Anodizing	TZ	0.633	0.811	**	6.93 ± 0.4
Coatings	Ti64 + TiB ₂ microcomposite	IR	0.218	0.282	6.16 ± 0.27	5.45 ± 0.67
	Al-Mg-B nanocomposite	NAMB	0.054	0.075	*	7.1 ± 0.24
	Diamond like carbon coating	DLC	0.06	0.082	4.14 ± 0.46	4.56 ± 0.10
	TiN coating	TiN	0.118	0.153	7.1 ± 0.36	10.29 ± 0.47
	CrN coating	CrN	0.056	0.07	4.61 ± 0.11	7.1 ± 0.24
	Cu-Ni-In coating on BB	CuNiIn-BB	8.17	10.26	**	**
	Cu-Ni-In coating on Ti64	CuNiIn-Ti64	11.46	14.5	**	**
Mechanical treatments	Shot peening Ti64	SP	3.98	5.03	**	**
Duplex treatments	Oxygen diffusion + shot peening	OD + SP	2.57	3.23	**	**
	Nitriding + shot peening	NR + SP	2.14	2.69	**	**
	Carburizing + shot peening	CR + SP	2.09	2.65	**	**

Notes:

* Invalid data due to indentation fracture. There was a relatively soft substrate under a hard, thin coating

** Surface was too rough to measure the indent dimensions accurately

Pass-Fail Criteria for Lubricated Tests. ‘Pass’ criteria for lubricated ASTM G133 Proc. B(Mod) tests were based on a friction coefficient of less than 0.15, and a stable and non-vibratory behavior throughout the entire duration of the run. Some tests vibrated or squealed so badly, that a complete test could not be completed. Others that failed had severe wear damage or wore so quickly that the test had to be stopped to avoid damaging the specimen holders.

Non-lubricated wear of baseline couples. Table 6 lists the wear volumes of the flat coupons and the balls for non-lubricated baseline tests (ASTM G133, Proc. A). This geometry (a concentrated point load oscillating on a surface) typifies a worst case scenario because the contact stress is high and a stable lubricant film cannot form as the sliding speed rises and falls. From Table 6, the total wear volume for Ti64 against 52100 steel is about 30% greater than that of the baseline bronze alloy against the same ball material. This performance is unacceptable.

Table 6. Wear Volumes (V_w) from Non-lubricated Tests

Sliding pair (flat specimen / ball)	V_w , flat specimen (mm^3)	V_w , steel ball (mm^3)	Total wear (mm^3)
Ti64 / 52100 steel	1.16	0.008	1.171
CDA 932 BB / 52100 steel	0.897	0.0017	0.899

Lubricated friction and wear results for Task 1. Figure 7 summarizes the frictional data from the Task 1 screening tests using the conditions shown in the right-hand column of Table 4. In addition to a friction coefficient criterion of < 0.15 , unstable sliding or excessive wear behavior further reduced the number of candidates selected for Phase 2.

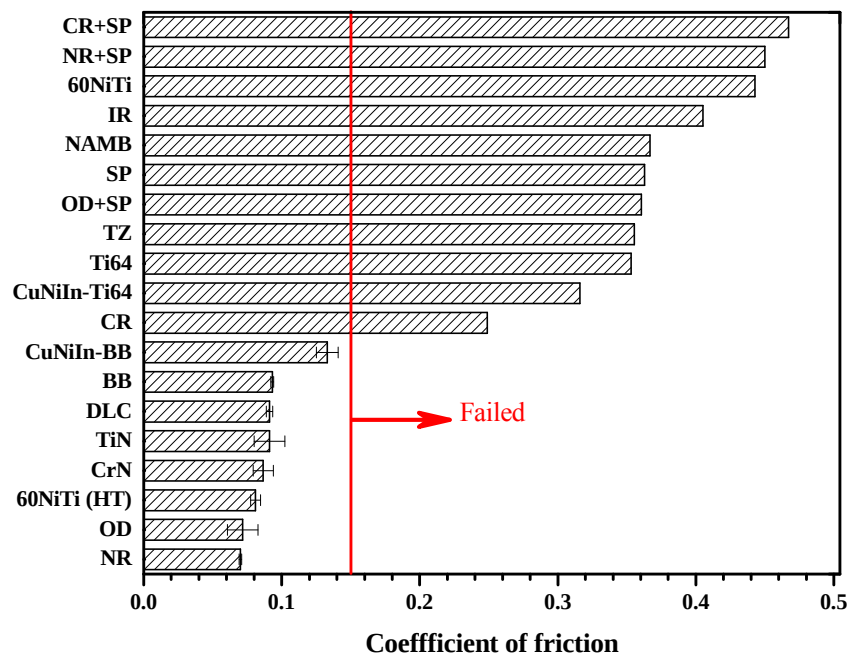


Figure 7. Average steady-state friction coefficients for candidate surface treatments under lubricated conditions, including data for the baseline bronze (BB) which passed, and the bare Ti64 which failed. Treatment code designations may be found in Section 9 and descriptions in Appendix B.

Figure 8 displays the three-test average wear volumes of both coupons and ball materials subjected to lubricated tests in which all three repeated tests ran to completion without the need for premature termination. The wear volumes of the coupons are presented on the right hand X-axis and the wear volume of the balls is plotted on the left hand X-axis. Note that the split axis uses different length scales on the left and on the right.

Of the coated specimens, CrN coated Ti64 showed the least amount of wear, followed by DLC coated coupons. Of the heat treated specimens, OD treated Ti64 coupons showed the least amount of wear. Tests with the Cu-Ni-In electro-coated BB (CuNiIn – BB) specimen produced the largest amount of wear on both the flat coupon and the balls, but the coating is soft and is intended as a sacrificial, running in coating. Among Ti alloys, the heat-treated 60NiTi custom-cast alloy (60NiTi + HT) shows the most wear on both the flat coupon and the ball. Since the wear volume of the coupons is orders of magnitude larger

than the wear volume of ball specimens, the ranking of the treatments remains the same even when total wear volume of both the counterfaces is considered. Photomicrographs of wear specimens and cross-sections thereof may be found in Bansal et al. [10] and are not reproduced here.

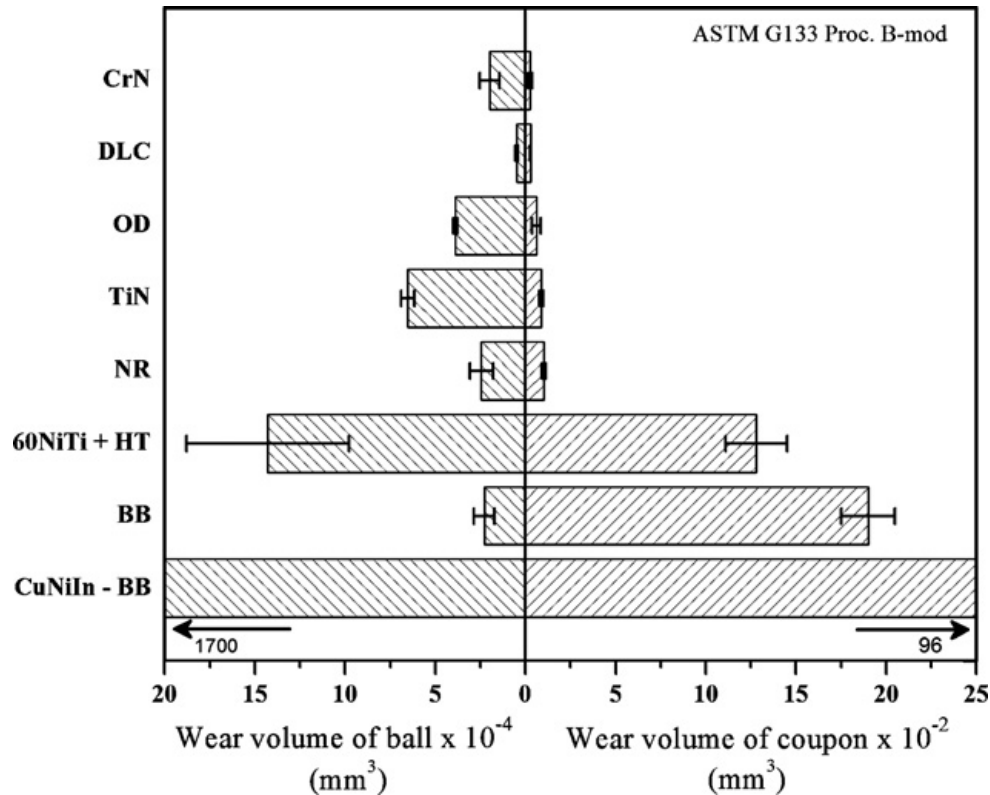


Figure 8. Comparison of the wear volumes of the ball (to the left of the center) and the flat coupon (to the right of the center) from [10].

Effects of Mechanical Working in addition to Diffusion Treatments. Additional studies on the tribological effects of surface treatments that combined mechanical working with diffusion treatments were performed early in 2012 [11]. Those experiments also used ASTM G133 (Procedure A) with parameters listed in Table 4. No lubricant was used in order to emphasize the material effects. Furthermore, two different slider materials were used: 440C stainless steel and silicon nitride (NBD 200, St. Gobain-Norton).

The diffusion treatments (done prior to mechanical treatment) were:

- Carburizing
- Nitriding
- Oxygen diffusion

The mechanical treatments applied to the coupons were:

- Shot peening
- Planishing

Shot peening of Ti64 is used extensively to improve its fatigue strength. It introduces residual compressive stresses into the surface. The shot peening in this work was done by a commercial facility (see Appendix A) using the following parameters: 230H cast shot with 0.008-0.012" A (Almen intensity) and coverage of 125% . Planishing is a process of repetitive impact with a rounded hammer. It is typically used for decorating or forming sheet metal, or for repairing damage in sheet metal such as in auto repairs. In the current case, planishing used a rounded-end (ball peen) hammer weighing 586 g, and with radius of curvature of approximately 25 mm. Starting with a titanium coupon polished with 600 grit wet SiC paper, hundreds of impacts were made to work harden the surface. After planishing, the Rockwell hardness of the Ti64 surface had risen from HRC 37.7 to HRC 48.

Table 7 lists the arithmetic average surface roughness (Ra) and ten-point height (Rz) of the pre-test coupons in various treated conditions, including the bare Ti64 substrate. Since some of the surface treatments were only tens of micrometers deep, it was not felt to be practical to attempt to final-polish the specimens after applying the combined surface treatments, but rather to leave them as-is. This may have affected the wear-in characteristics of the coupons, but in most cases, the wear rate would have quickly worn through the initial roughness.

Table 7. Pre-Test Roughness of Bare and Treated Titanium Alloy Specimens
(Ra = arithmetic average roughness, Rz = ten point height, 6 measurements)

Treated Condition	Ra (μm) Average (std. dev.)	Rz (μm) Average (std. dev.)
Bare Ti64	0.213 (0.03)	2.07 (0.36)
Planished only (PL)	0.195 (0.02)	1.60 (0.35)
Shot peened only (SP)	2.953 (0.25)	15.87 (1.03)
Nitrided + shot-peened (NR + SP)	1.617 (0.08)	8.35 (0.51)
Oxygen diffused + shot-peened (OD + SP)	1.955 (0.24)	10.98 (1.60)

Subsurface Microstrain Characterization. In addition to surface finish measurements and hardness testing, the degree of surface damage imparted by the various treatments was investigated using X-ray diffraction (XRD) line broadening analysis. Diffraction peaks broaden due to crystallite size and microstrain [14]. Smaller crystallites cause broader peaks, but the effect is negligible for crystallite sizes greater than approximately 100 nm. Microstrain includes any effect causing a non-uniform, local variation in lattice spacing, such as composition gradients and stacking faults. For a ductile metal such as Ti, the largest source of microstrain broadening is dislocations. Size and strain broadening effects may be distinguished by their dependence on the diffraction angle. The peak areas remain constant, and therefore peak broadening is accompanied by a reduction in peak height.

Diffraction patterns were collected with a PANalytical X'Pert Pro diffractometer in Bragg-Brentano geometry with Cu K α radiation ($\lambda=1.5406$ Å). With Cu radiation, the penetration depth (99%) into Ti64 is estimated to be 5-16 μm, depending on incident angle. Because the typical depth of the surface-affected zone was on the order of 100 μm, the measurements were restricted to the affected zone. The data were fit using the LeBail method with the pseudo-Voigt profile function and using the Cagliotti equation to fit the peak widths. The microstrain (ϵ) may be determined from the peak widths as in the following equation $\epsilon=B_{\text{strain}}/(4\tan \theta)$, where θ is half the diffraction angle and B_{strain} is the peak broadening

due to strain obtained by removing the instrumental and size broadening from the total measured peak widths [14].

A silicon standard was used to remove the instrumental effect from the total broadening. Preliminary analysis indicated negligible size broadening, and therefore it was assumed to be zero. Additionally, the microstrain was assumed to be isotropic. A more in-depth analysis could account for anisotropic broadening arising from preferential slip directions [15], but is beyond the scope of this work. The microstrain was calculated from the peaks of α -titanium, which was the major phase in all samples. Traces of β -titanium were observed in the non-treated and planished samples, and traces of nitrides (namely Ti_2N and TiN) were observed in the nitrided samples.

The microstrain values are shown in Figure 9. The Ti64 sample without any additional surface treatment beyond the as-received condition has the lowest value, as expected. Of the two mechanical surface treatments, shot peening resulted in a much higher level of microstrain than planishing. Diffusion treatments with oxygen and nitrogen alone contribute to the measured microstrain, likely due to chemical inhomogeneity in the metal and/or the metal-oxide/nitride interface stresses. The combination of diffusion treatment with shot peening gave similar results to shot peening alone. The increase in microstrain arising from diffusion treatment may be offset by a reduction in microstrain arising from dislocations, due to shielding of the underlying metal due to the surface oxide or nitride.

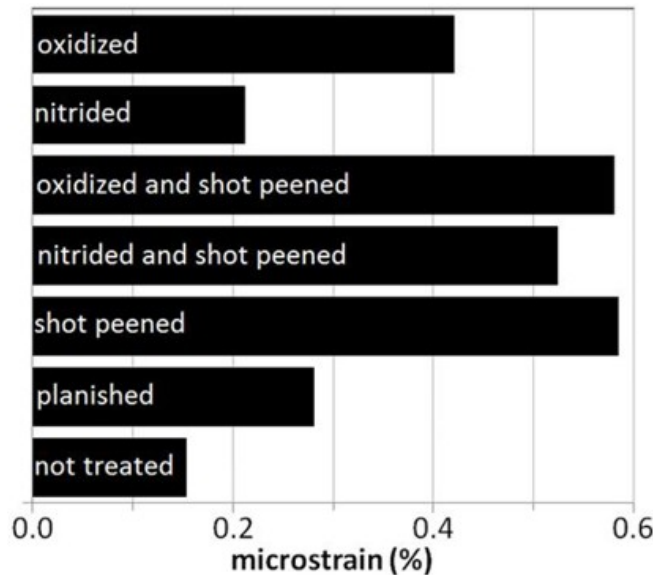


Figure 9. Comparison of subsurface microstrains for various treatments of Ti64.

Friction and Wear Results. The most pronounced effects of combined treatments were observed in the frictional behavior, as exemplified by Figure 10. Without treatment, the kinetic friction coefficient of the materials couple rose quickly to about 0.4-0.5. With the combined surface treatment, there was a delay, and a lower friction initial period preceded the transition. The choice of ball material also seemed to affect the duration of the lower friction period, with the 440C ball delaying the transition to higher friction by about 3 meters in sliding distance relative to the ceramic ball.

Since friction force times distance equals frictional work, the longer a sliding couple remains at low friction, the less energy is available to do work on the materials – i.e., to produce wear. Therefore, one

would expect less total wear for the tests in which there was a lower friction period. The data in Table 8 show that this effect was relatively minimal. In all but one case (NR+SP), the low friction period was less than 10% of the total sliding distance. In the case of (NR+SP) coupons rubbed on 440C steel was there a discernible effect of the low friction period on reduced wear volume.

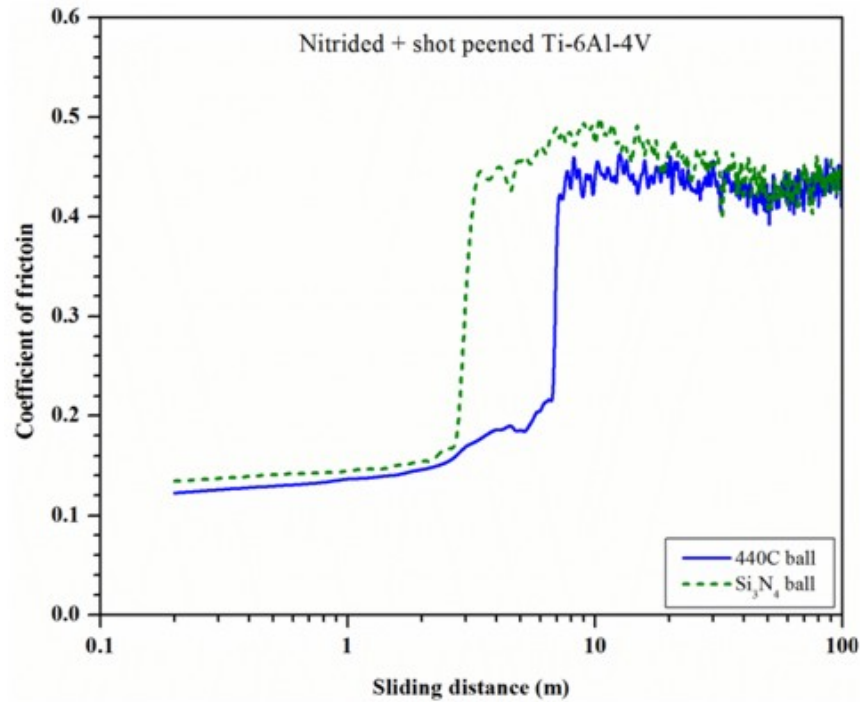


Figure 10. Friction transitions during initial period of sliding of NR+SP coupons against two ball materials. In both cases the initial and final friction coefficients were about the same.

Table 8. Friction Transition Distances and Wear Volumes of Surface Treated Ti64 Coupons

Slider material	Treatment	Test	Sliding distance to friction transition (m)	Wear volume, flat specimen (mm ³)
440C st steel	PL	1	none	0.986
		2	none	1.121
	SP	1	none	0.985
		2	none	1.103
	OD + SP	1	6.8	0.901
		2	4.6	1.012
	NR + SP	1	8.1	0.802
		2	23.2	0.823
Silicon nitride	PL	1	none	1.124
		2	none	1.025
	SP	1	none	1.185
		2	none	1.143
	OD + SP	1	0.4	1.066
		2	0.8	1.050
	NR + SP	1	2.6	1.062
		2	1.2	1.097

The primary conclusions and observations of studies of combined treatments are:

- The choice of a counterface material, whether 440C stainless steel or bearing-grade silicon nitride, had little effect on long-term friction trends. However, for combined diffusion treatments plus mechanical working, the 440C balls displayed longer initial low friction period before transitioning to the higher friction typical of non-treated Ti alloy.
- The choice of counterface material had no evident effect on the wear of shot peened or planished (mechanically-worked) Ti64 surfaces.
- Silicon nitride balls displayed transferred material from Ti64 from the flat coupons, but there was no significant difference in the wear amount from one treatment to another.
- The combination of nitriding with shot peening yielded the lowest wear results when sliding against 440C.
- Planishing and shot peening added to the near-surface microstrains in the non-treated Ti64 specimen.
- Oxygen diffusion and nitriding treatments increase the microstrain, but did not raise the microstrain levels as high as those from mechanical working alone.
- No obvious gradients in hardness were recorded beneath the wear track surface for sliding against either of the counterface materials, silicon nitride or 440C balls although there were some indications of a mechanically deformed layer.

In general, the sliding wear process for titanium alloy surfaces is so problematic that shallow diffusion treatments have little effect other than to prolong the transition to severe wear under non-lubricated sliding conditions. Additional details of this study, including microscopy, may be found in Ref. [11].

4.3 Development of the Variable-Load Bearing Test (VLBT) Apparatus

A programmable, variable loading, rotating cylinder-on-flat apparatus was designed and built to study the friction and wear of several non-ferrous alloys and surface treatments under variable loading conditions. The basic configuration of the VLBT is shown in Figure 11. A close-up is shown in Figure 12. The load is applied by a displacement-controlled, stiff ball screw that compresses a calibrated spring at the end of a primary load arm. A computer program varies the actuator movement according to a preprogrammed load spectrum, and hence the applied load on the stationary specimen that is pressed against a rotating cylinder of alloy steel. The load spectrum was based on a polar plot of the forces on a big-end bearing in an automobile engine that was published by Taylor [9]. The variations in the load with degrees of crank shaft rotation were scaled to fit the load range of 20-100 N that can be applied by the VLBT. To study frictional effects on the bearing surfaces at each load in more detail, the period of a single cycle was stretched to 120 s, whereas the whole sequence of load changes might take place in a fraction of a second in an actual engine operating at 1000-4000 rpm. The standard 120-second load profile for the majority of tests presented was comprised of the following sequence of forces and times: 36 N (10 s), 60 N (10 s), 80 N (10 s), 50 N (15 s), 75 (10 s), 100 N (10 s), 50 N (10 s), 75 N (10 s), 50 N (15 s), 80 N (10 s), 60 N (10 s), 36 N (10 s). This profile was repeated for a total of ten times. An example of the first two loading cycles is shown in Figure 13 for a test on a bearing bronze specimen against steel.

The surface speed used for the current experiments was 0.50 m/s. A peristaltic pump (showed at the rear of the view in Figure 10) dispensed one droplet of 15W40 diesel oil (Valvoline Blue™ diesel oil) about every 14 seconds, resulting in a delivery rate of 11.8 ml/hr, or about 3.93 ml for a 20 minute test. Friction

force, normal force, and motor speed were sampled at a rate of 1000/s and the friction coefficient, averaged over 2 second intervals, was used to establish trends. The friction coefficient-load behavior during the last full test cycle was used to determine the post-running-in behavior for the various materials.

Typical wear scars on a bronze test specimen (CDA 932) against 8620 steel are shown in Figure 14. More testing details are provided in sections 4.3 and 4.4 of this report.

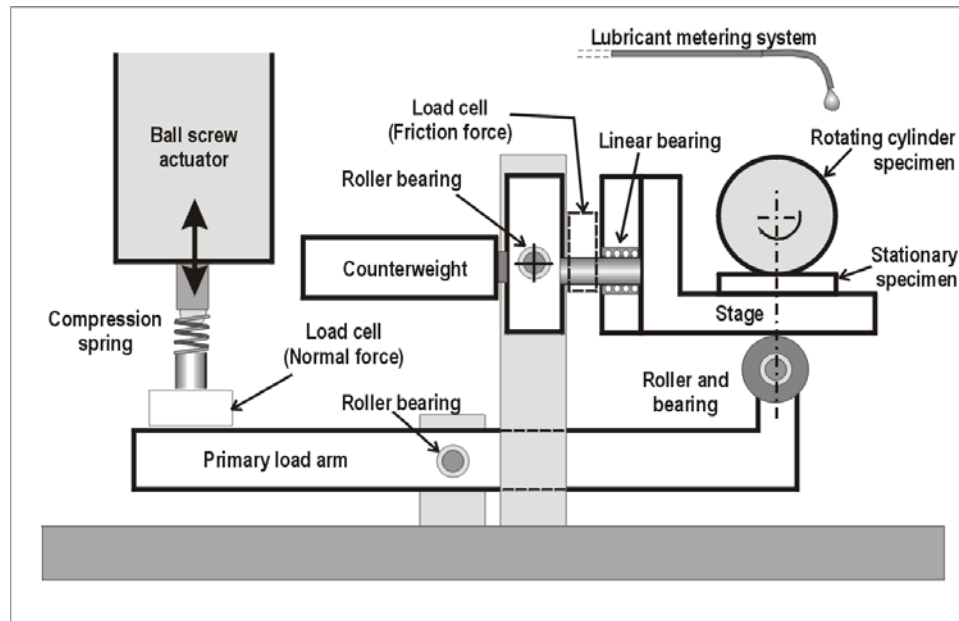


Figure 11. Schematic arrangement of the variable load bearing tester (VLBT).

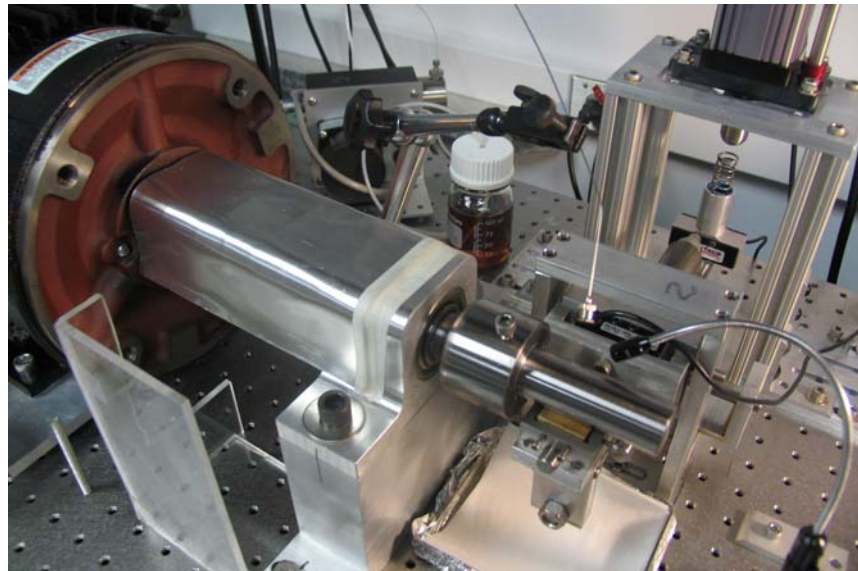


Figure 12. Close-up of the VLBT showing the rod specimen (lower right), the flat specimen pressed up underneath the rod, the stainless steel lubricant delivery tube (above the contact area), the load actuator (upper right), the oil pump (upper left, at rear) and the motor drive train (left).

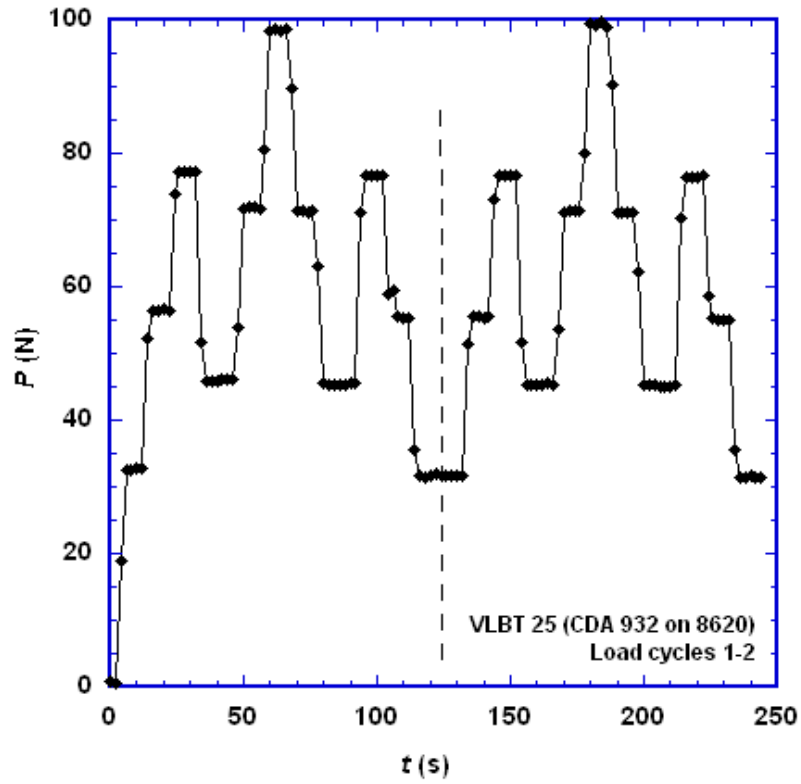


Figure 13. Load versus time for the first two cycles of a standard VLBT test.

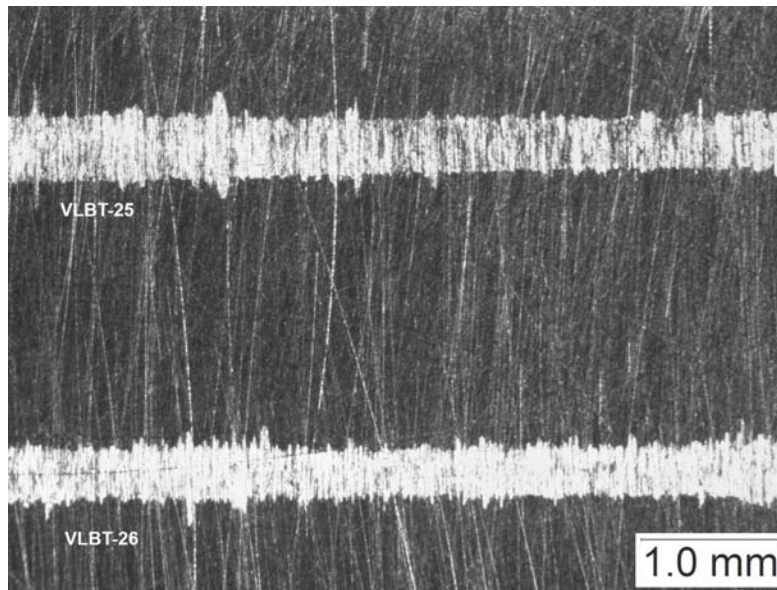


Figure 14. Two typical wear scars on a flat specimen (bronze) against alloy steel, produced on the VLBT. It was run for ten variable loading cycles lubricated with diesel engine oil.

4.4 Phase 2 Test Results

These results are also described in the proceedings of the 19th International Conference on Wear of Materials, to be published in *Wear* journal [16].

4.4.1 Wear comparison between constant and variable load tests. In early experiments with the VLBT system using Ti-6Al-4V on 8620 steel, a variable loading test having equal sliding distance and speed as constant load test (the constant load was the average of the variable loads) were compared. Figure 15 contrasts the friction coefficient behavior.

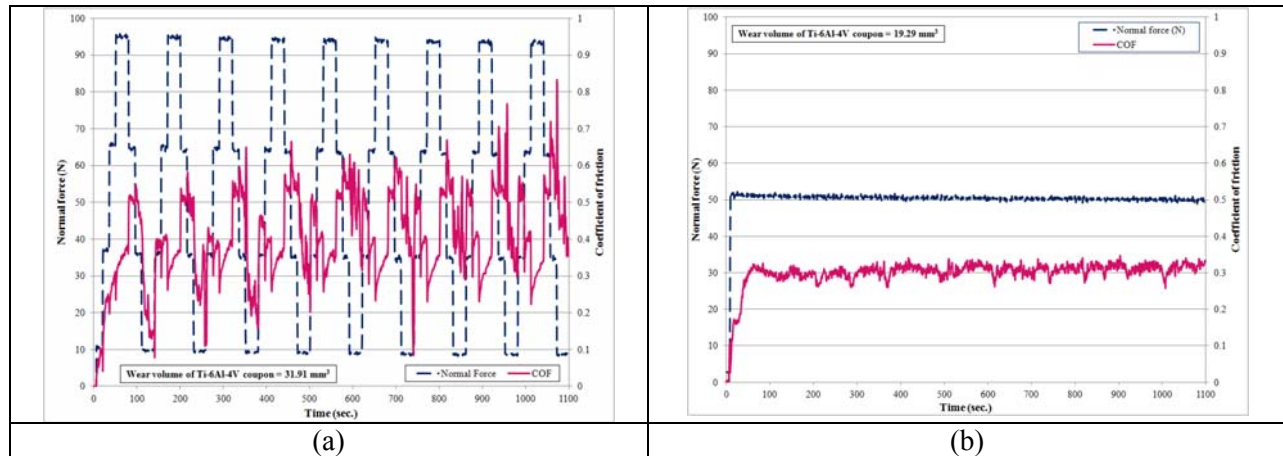


Figure 15. Comparison of load profiles and friction coefficients for a variable load test (a) and constant load test (b). In (a) the load is represented by dashed line and the friction coefficient solid lines. In (b) the friction coefficient is the lower of the two curves. The sliding speed, duration, and average normal force was the same for both tests. (Vertical scales are the same on both figures.)

As shown in Table 9, the wear volume of the flat specimen after testing was considerably larger for the variable load test than for the constant load test. This has significant implications for laboratory simulation of engine components because it is not sufficient to simply choose a nominal operating load for screening materials and lubricants. The load should vary similarly to the operating conditions.

Table 9. Comparison of Wear Volumes for Diesel Oil Lubricated Tests of 8620 Steel Against Ti-6Al-4V

	Wear volume of flat coupon, $V_{w,f}$ (mm ³)
Variable load test	31.91
Constant load test	19.29

4.4.2 Running-in response. In determining the length of standard tests for this work, it was desirable to ensure that the surfaces were sufficiently worn-in so as the data obtained would be representative of steady-state operating conditions. During the first few experiments, it was clear from the friction record that all test combinations would quickly run in to a steady state condition within less than one complete

loading cycle. For example, the bronze ran in distinctively during the first $\frac{1}{2}$ loading cycle (as shown in Fig. 16). After the initial steep friction transient, the friction coefficient versus normal force at steady-state was relatively stable (Fig. 17). By the 10th cycle, the friction coefficient for the bronze was quite stable, despite the changing loads (Fig. 18). As a result of initial experiments like these, the tenth cycle of spectrum testing was used as the standard to evaluate the friction coefficient versus load behavior of the candidate surface treatments and coatings.

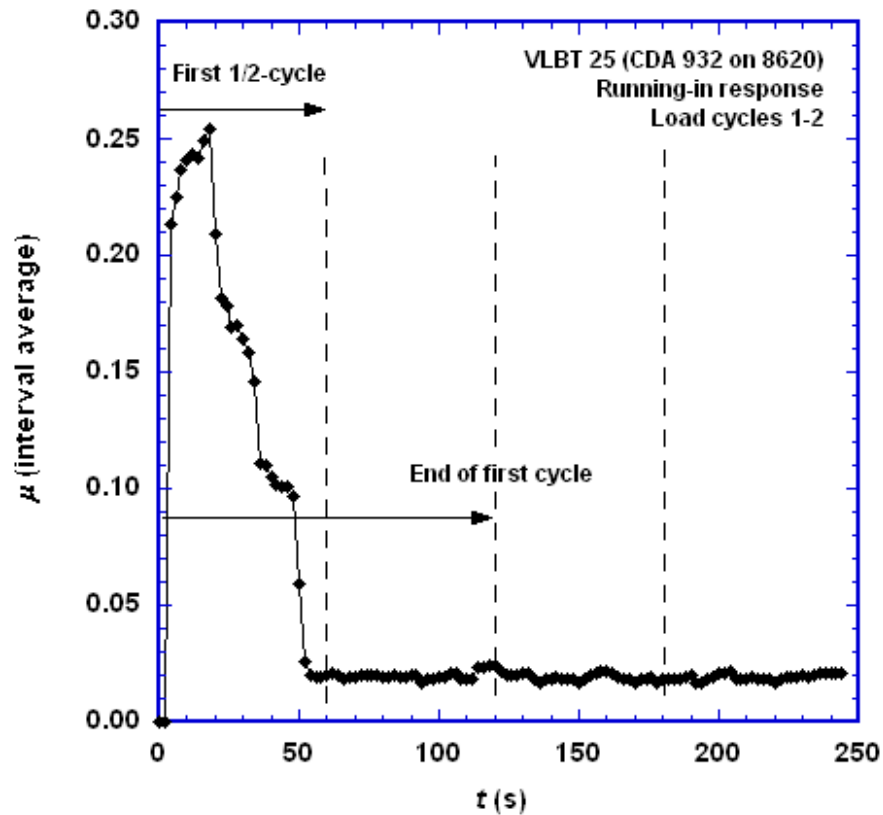


Figure 16. Running-in of bronze on 8620 steel during the first half cycle of the load spectrum.

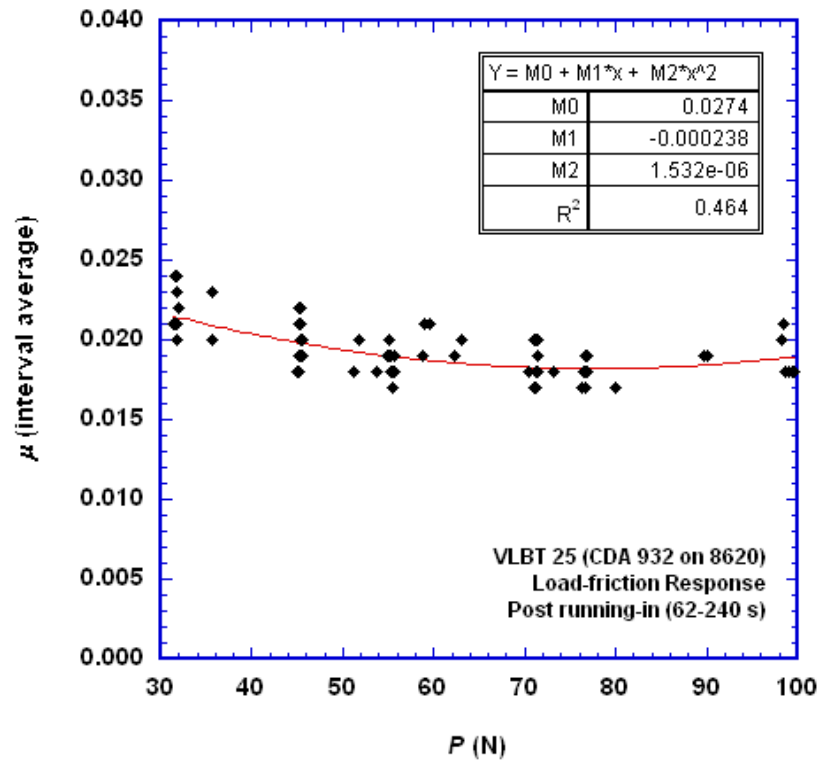


Figure 17. Post running-in friction versus normal force for the same run as shown in Fig 14.

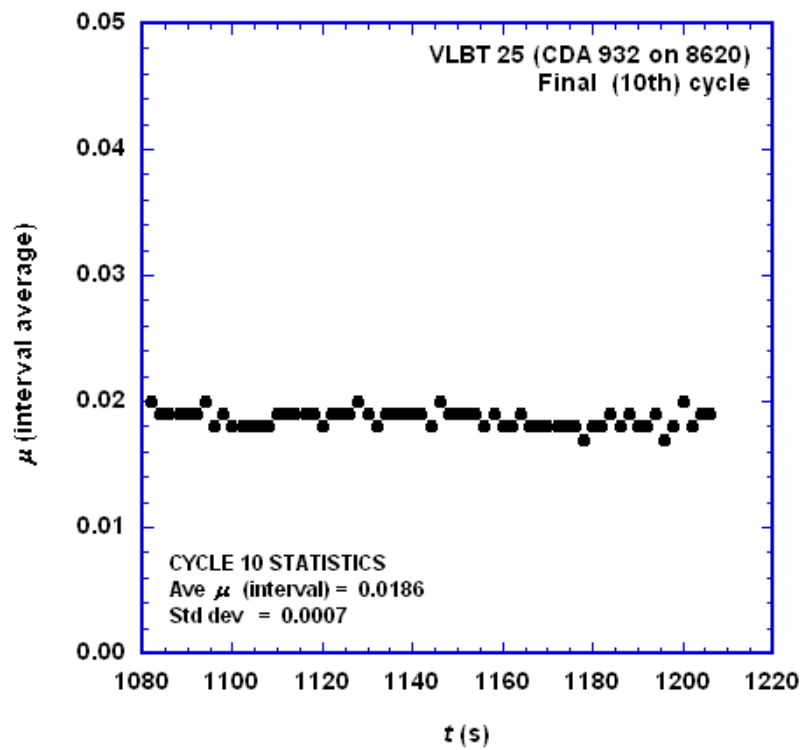


Figure 18. Friction coefficient versus time during the tenth cycle of spectrum loading of bronze on steel in diesel oil.

The film thickness/roughness ratio (Λ , see Fig. 4) is a useful descriptor of the lubrication regime. The numerator (h) is the lubricant film thickness in the bearing, which is usually calculated from lubrication theory [15] rather than measured. The denominator of that parameter is related to the root-mean-square roughness (σ_o) of the two opposing surfaces, thus:

$$\sigma_o = \sqrt{\sigma_1^2 + \sigma_2^2} \quad (1)$$

During running-in (wear-in), the roughness of the softer of the mating surfaces tends to change more than the harder surface. If the process is one of asperity truncation, then the surface becomes smoother during wear-in. More often, both surfaces change somewhat during wear-in. However, if the load is high enough, the wear becomes more severe than simply truncating the sharper peaks on initial the surface finish and the post wear-in roughness increases.

Λ is affected by the wear-in of opposing surfaces, and friction coefficient tends to reach a minimum when $\Lambda \geq 3.0$. Considering that a high lambda implies a relatively thick lubricant film and separation of solid surfaces, then the effects of roughening or smoothing during wear-in can be demonstrated for several example cases, which start with both surfaces having $\sigma = 0.5\mu\text{m}$.

The effects of changing either σ_1 or σ_2 or both are plotted in Figure 19, assuming a constant film thickness of $1.0\mu\text{m}$.*

Case 1: σ_1 decreases (wears smoother) to $0.01\mu\text{m}$ while σ_2 does not change

Case 2: σ_1 and σ_2 both decrease (wear smoother) to $0.01\mu\text{m}$

Case 3: σ_1 roughens up to $2.0\mu\text{m}$ while σ_2 does not change

Case 4: σ_1 and σ_2 both roughen up to $2.0\mu\text{m}$

It is presumed that some running-in occurs on every VLBT run; therefore, it would be presumptuous to use the initial surface roughness to estimate Λ at post run-in steady-state conditions.

* Measurement of actual film thicknesses in connecting rod bearings during engine operation is very difficult, so these values are usually calculated using one of several available theories. For example, Sun et al. [17] have calculated film thicknesses in internal combustion engine main bearings as a function of crank angle. The plot is shaped similarly to the triple-peaked behavior in Figure 11, except that the central peak is suppressed because the pressure in the film rises during the central spike, compressing the film. Values for the minimum lubricant film thickness range between 1.2 and $2.2\mu\text{m}$ from the calculations reported in [17]. Other work on connecting rod bearings by Ross and Slaymaker [18] reports minimum film thicknesses, during 720 degrees of crankshaft rotation, ranging between 1.1 and $8\mu\text{m}$ for a spark ignition engine running at 5250 rpm. Assuming $1.0\mu\text{m}$ for h in Figure 17 seems reasonable for the purposes of illustration.

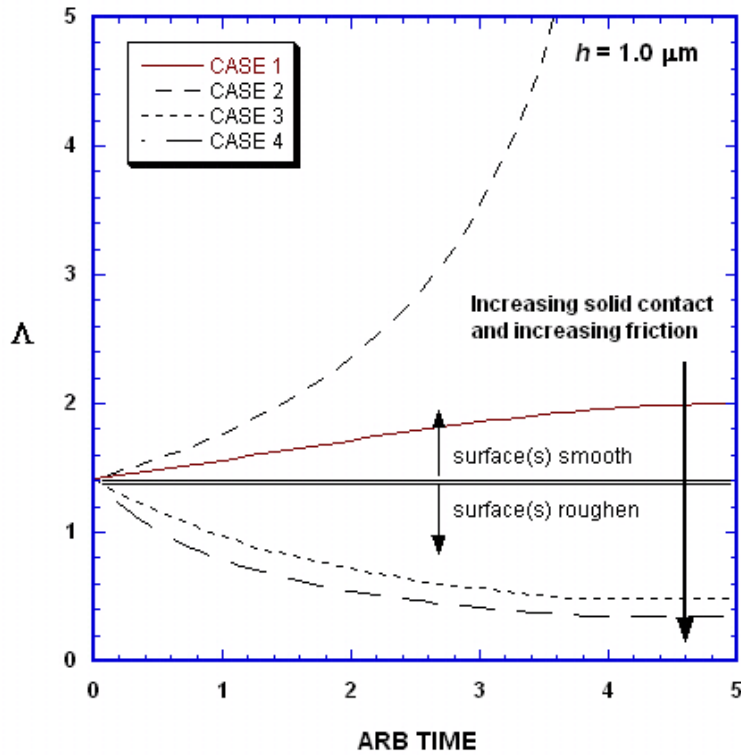


Figure 19. Effects of roughness on film thickness ratio as time passes (arbitrary time scale), given $h = 1.0 \mu\text{m}$.

4.4.3 Summary of friction data for various surface treatments. From the standpoint of lubrication regimes, VLBT operating point in essence moves up and down the Stribeck curve (e.g., Fig. 4) as the applied load cycles through its programmed variations (as indicated in Fig. 13). To examine that relationship friction coefficient data for the tenth cycle of each test were plotted not as a function of time, but rather versus the inverse load (P^{-1}) during the final cycle of the run. In each case, two experiments are plotted in the same figure to account for the repeatability of the data. In examining the plots in Figures 20-28, one is struck by the similarity of these plots to the bottom portion of Figure 4 which is the beginning of full film conditions in which friction tends reach its minimum.

Third-degree polynomial curve fits were applied to all the data presented in Figures 20-28 (see Appendix B for raw data). Two representative values, the friction coefficients corresponding to 30 N and at 100 N were calculated from those curve fits. Those results are summarized in Figure 29 to illustrate the maximum, minimum, and range of friction coefficient values obtained under steady-state lubricated sliding conditions. The reference material (Brnz) had the smallest range but the carburized surface, the CrN coated surface, and the HVPC coated surface all had lower friction levels during that cycle.

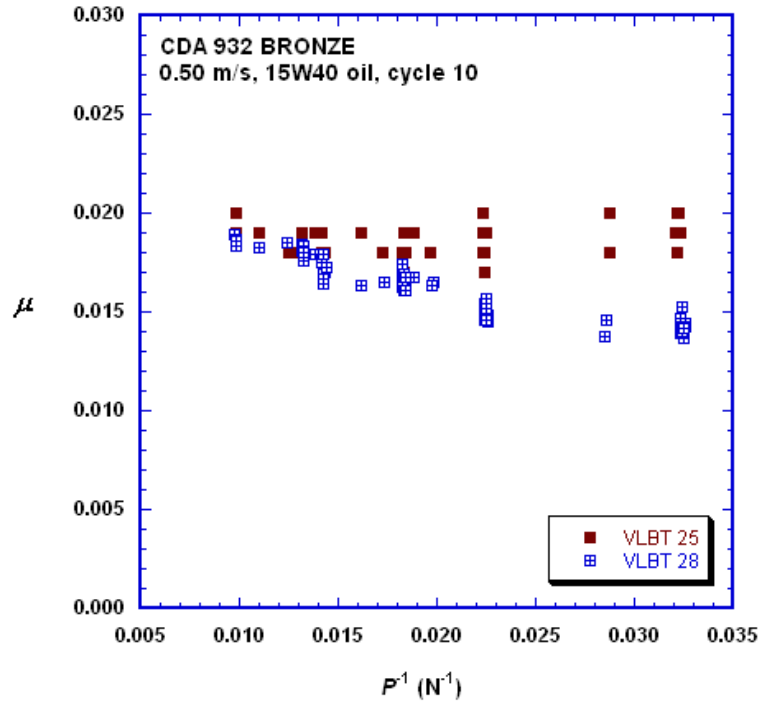


Figure 20. Friction coefficient versus inverse load for bronze.

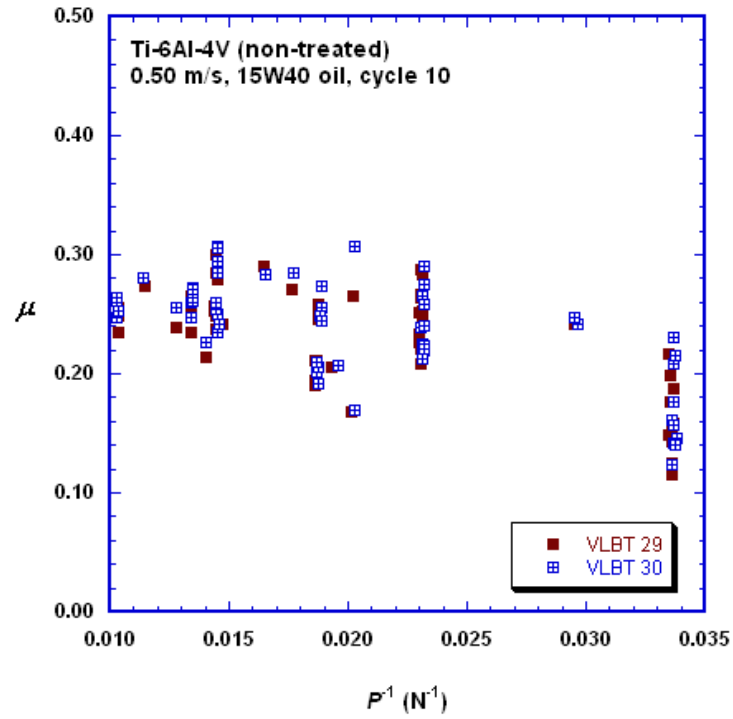


Figure 21. Friction coefficient versus inverse load for Ti64. Friction levels are associated with the boundary lubrication regime, and hence a different shaped curve than other combinations.

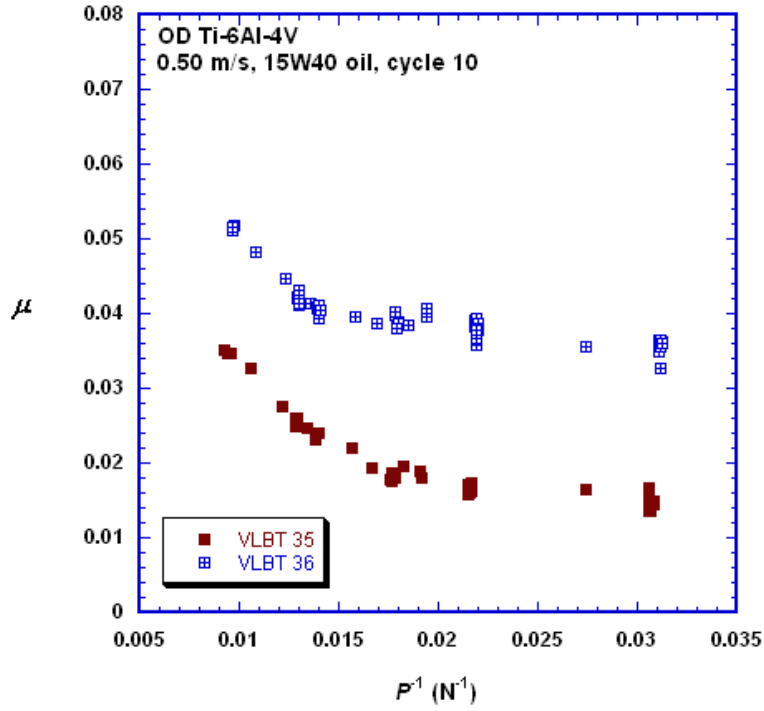


Figure 22. Friction coefficient versus inverse load for OD Ti64. The reason for the off-set curves is not known, because tests run both before and afterward suggest that the response was real and not a calibration artifact of some kind.

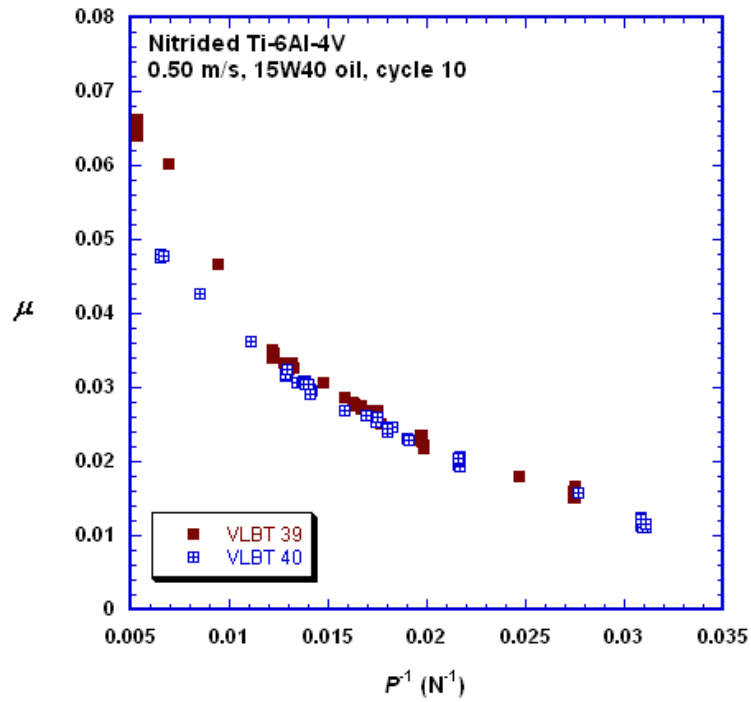


Figure 23. Friction coefficient versus inverse load for nitrided Ti64. Results at loads below 100 N (that is, $P^{-1} > 0.01$) were very repeatable. High load data tended to differ between runs.

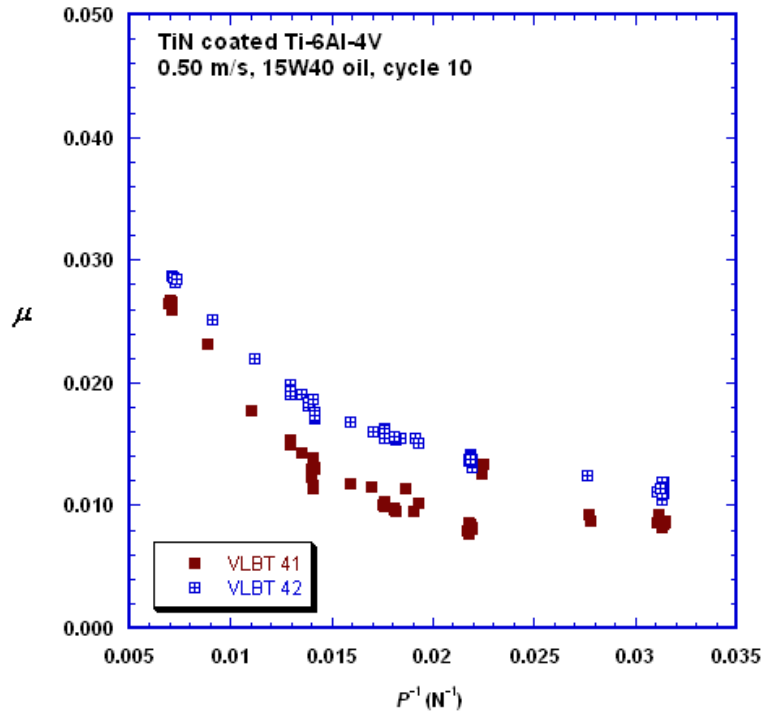


Figure 24. Friction coefficient versus inverse load for TiN coated Ti64.

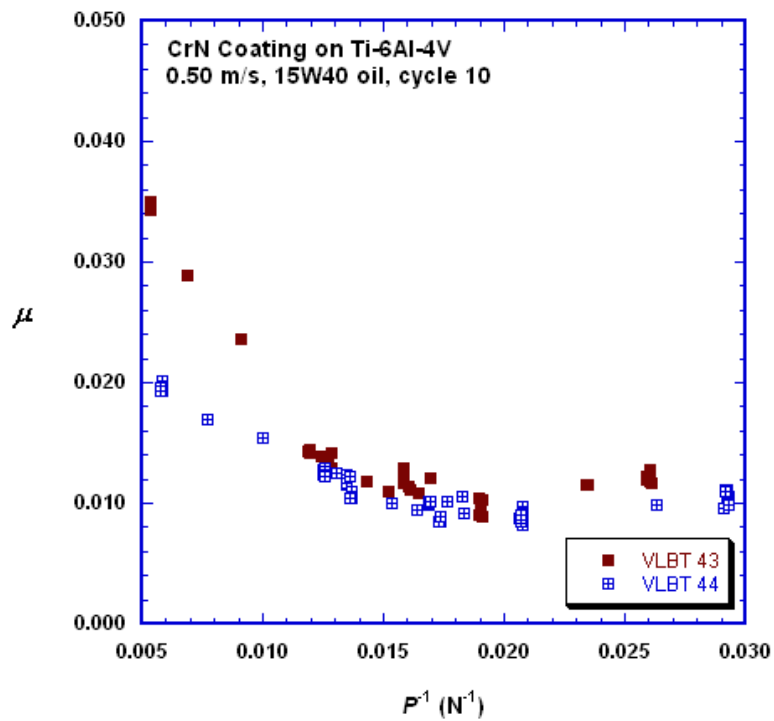


Figure 25. Friction coefficient versus inverse load for CrN coated Ti64

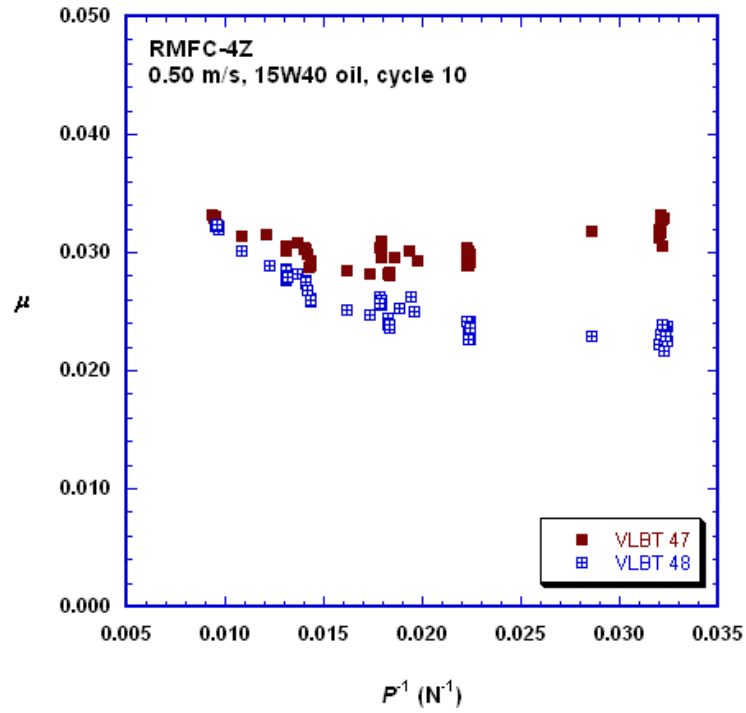


Figure 26. Friction coefficient versus inverse load for the polymer-filled anodic treatment (TZ).

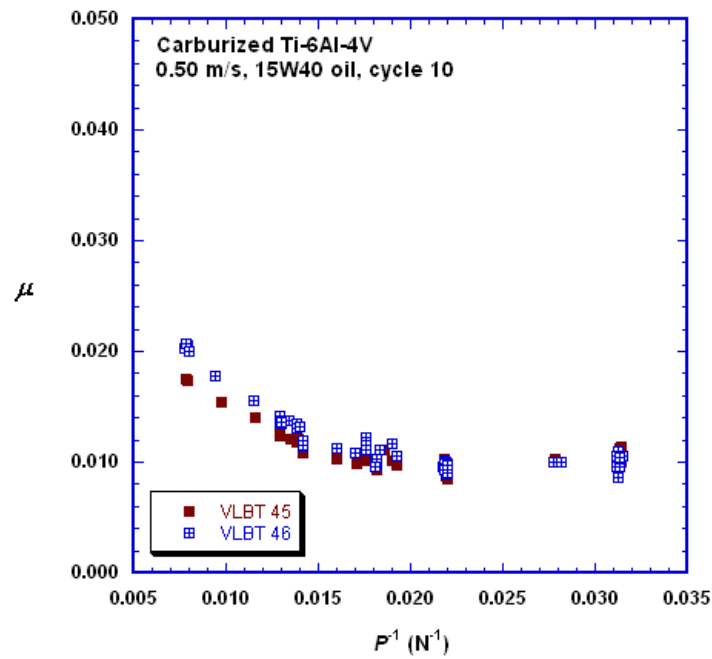


Figure 27. Friction coefficient versus inverse load for carburized Ti6. Friction coefficients were among the lowest of those measured in these experiments.

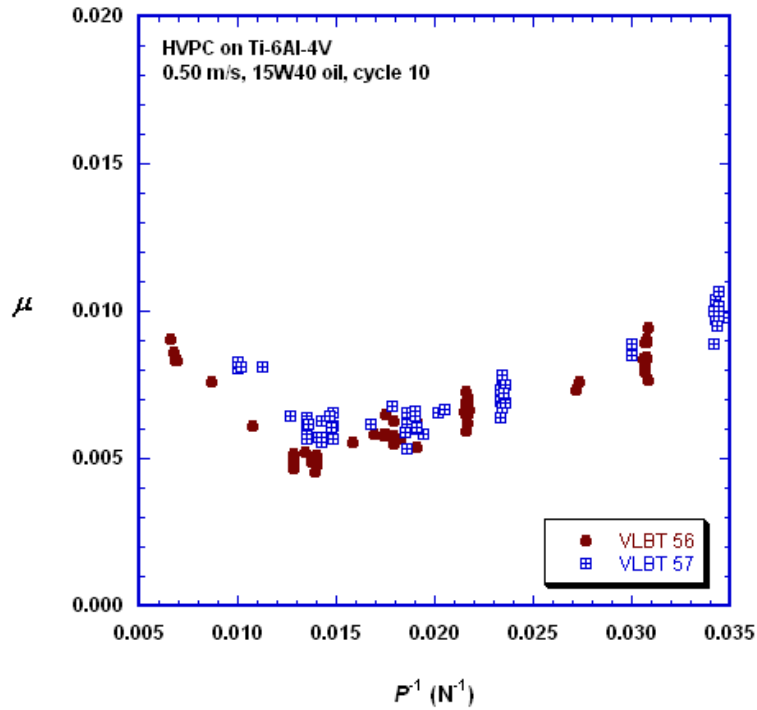


Figure 28. Friction coefficient versus inverse load for the high velocity powder consolidated coating from Pennsylvania State University.

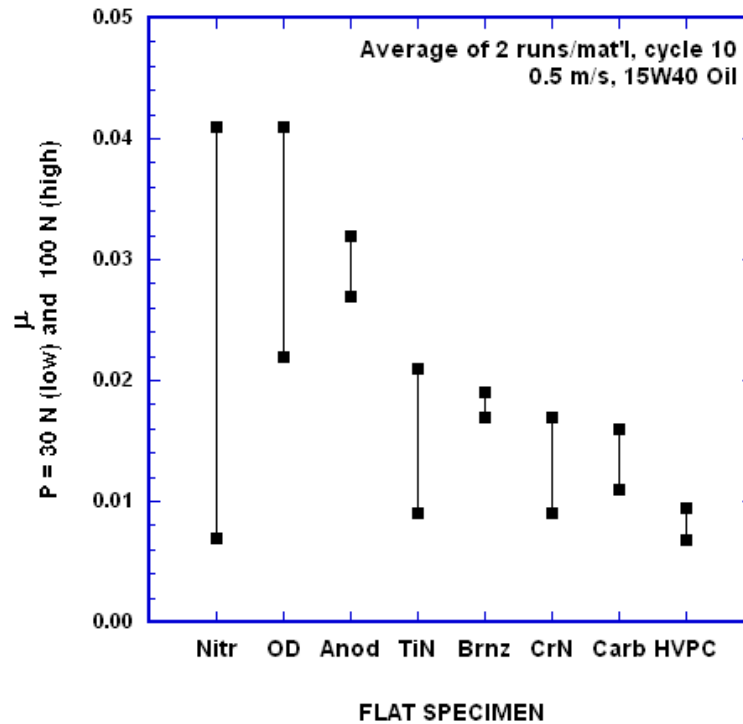


Figure 29. Summary of cycle 10 friction data showing the maximum, minimum, and range of friction. The reference material (Brnz) had the smallest range but three surface treatments had lower friction during the cycle.

4.4.4 *Wear surface examination.* Operating so closely to the mixed- and thick-film regimes produces little measurable wear damage and it is impractical to calculate an accurate wear rates for any of the test specimens except the most grossly worn Ti64 which is not of practical interest. However, microscope examination of the contacting surfaces distinguished some differences between the various surface treatments studied in Phase 2. Since the friction coefficients were not equal for the various material combinations, we must assume that the tribological behavior was not controlled by lubricant properties alone. In fact, the presence of surface features, to be described subsequently, indicates that the materials do play a role and that they do experience different degrees of wear despite a nominally mixed-film regime.

General Observations. General observations of the wear and surface contact features on the flat VLBT specimens are summarized in Table 10. Since the tests were mainly operating in a lubrication regime that allowed only rare solid-solid contact, it was not possible to quantitatively measure the wear volumes or estimate a wear rate. This practical consideration makes the prediction of wear lives for well-lubricated systems extremely difficult.

Table 10. Wear Features on Flat Specimens

Treatment	Test no. VLBT-	Observations of wear
Bare CDA 932	25,26	Even abrasion across the length of the contact zone.
Bare Ti64	29,30	Deeply scored wear scars with significant plowing and abrasion, suggestive of severe metallic wear. Friction levels consistent with boundary lubrication or starved boundary lubrication.
OD	35, 36	Pebbled reddish surface with numerous plateaus of bright, polished material and pitting indications as well. Suggests non-uniform wear and the potential production of hard particles removed by a mechanical pitting process. No abrasion indications like those on the bronze.
Nitrided	39, 40	Light wear scar with a few localized zones that displayed sharp scratches. Suggestive that the wear was generally mild except where an occasional debris particle could detach and run through the contact producing a scratch.
Carburized	45, 46	Faint, barely visible scar that did not run completely across the flat specimen, a few scuffed looking patches, and a few areas where some sharp scratches ran across the scar. Fewer scratches than for the nitride specimen.
TiN coated	41, 42	Very difficult to see any trace of wear, by eye or in the microscope.
CrN coated	43, 44	Very difficult to see any trace of wear, by eye or in the microscope.
Anod/coated	47, 48	There was a ground appearance to the unworn surface, running transverse to the sliding direction. Light polishing occurred in certain areas and the fibrous microstructure of the composite was revealed in those areas. A few more prominent scratches seem to run across the composite in the wear scar area and seemed to scratch continuously through the composite material all the way from the inlet side to the outlet edge of the scar.
DLC coating	(note 1)	Wore through to the Ti64 substrate, leaving jagged edges on the wear scar.

Note 1. Initial tests of this material displayed cracking through the deposit and rapid wear-through to the substrate, so this material could not be fairly evaluated and was eliminated from further testing.

4.5 Assessment of Surface Engineering Approaches for Ti-Alloys

A connecting rod big end bearing or other unidirectionally-moving component requires maintaining a regime of thick-film lubrication as much as possible to avoid solid contact and to enable the materials to withstand long periods of operation without excessive wear. If the majority of time of operation is spent with a well-established film of lubricant in the interface, then the materials need only sustain intermittent contact as the system starts up or shuts down. On the other hand, if the direction changes rapidly, as in a piston ring riding up and down a cylinder liner or the small end bearing of a connecting rod, then a stable lubricating film exists mainly during the mid-stroke portion of the cycle, and the materials play a more important role. The tests in Phase 1 were harder on the test materials than those in Phase 2 for that reason. Two of the surface treated Ti materials that failed Phase 1 (pin-on-flat tests) performed well in Phase 2 conditions (VLBT tests).

Consideration of the experimental data in the previous sections (reciprocating pin-on-flat (RPOF) versus unidirectionally rotating cylinder-on-flat (VLBT)) clearly demonstrates that a series of candidate materials can rank in a different order of merit when different friction and wear test methods are employed. In the case of reciprocating ball-on-flat tests, the contact pressure was in effect concentrated by the use of a spherical pin tip. Lubricant flow under such conditions also differed because flow around the sides and the direction of motion frequently reversed. Under the VLBT tests, the lubricant was fed into a converging wedge and the direction of motion of the test specimens was not reversed. Consequently, we screened the first series of candidate treatments (Section 4.2) in a different manner than the second set (Section 4.3-4). Results with respect to friction rankings are shown in Table 11.

Table 11.
Comparative Ranking of Surface Treatments by Two Different Tribo-Test Methods.

Flat Specimen Surface	Phase 1: RPOF (note 1)	Phase 2: VLBT (note 2)
High velocity powder consolidation (HVPC)	(note 3)	1
Carburized (Carb)	9 (note 4)	2
Chromium nitride coated (CrN)	4	3
Reference: CDA 932 (Brnz)	7	4
Titanium nitride coated (TiN)	5	5
Filled anodized layer (Anod)	12 (note 4)	6
Oxygen diffused (OD)	2	7
Nitrided (Nitr)	1	8
Ti alloy (Ti64)	11 (note 4)	9

Notes:

- 1) Ranked by lowest friction coefficient having the best ranking, and with greater than a friction coefficient of 0.15 considered 'failed.' Not all candidates from Phase 1 are listed because they were screened out prior to Phase 2. Therefore, some numbers are missing in the rank order
- 2) Ranked in considering both the lowest friction during the cycle, and the smallest range in friction coefficient (during a cycle than ranged from 30-100 N load)
- 3) The HVPC was not available for RPOF testing during Phase 1.
- 4) Did not pass Phase 1 criteria, but were tested in Phase 2 to determine if their behavior under different testing conditions would improve.

While the VLBT is probably a better simulation of a big end connecting rod bearing, the behavior of the various candidates to RPOF tests is a more rigorous test of the materials under boundary lubricated conditions. In that case, one needs to consider both results: the former as a well-lubricated test, and the latter as metric for transient, more severe conditions. From that point of view, the commercial CrN coating ranked high on both types of tests, although the HVPC coating also looks very attractive even though it was not tested under RPOF conditions in Phase 1.

Finally, these tests were conducted at ambient temperature. Conducting tests at engine temperatures could produce different results because: (1) the viscosity of the oil tends to decrease at elevated temperatures, leading to thinner fluid films and more solid contact, thus the role of materials properties would tend to become more important as temperature rises, (2) materials like the anodizing treatment with its polymer filler might not perform as well at elevated temperature due to filler softening, (3) the additives in formulated lubricating oils are activated by elevated temperatures and therefore, the chemical interactions, especially under boundary lubrication, could affect test results. It was found, for example, that the OD treatment for Ti favorably forms anti-wear films with formulated lubricating oils [19].

As a result of the foregoing discussion, certain additional lines of enquiry should be pursued:

- Considering the leading candidates, what effects do elevated temperature and used oil have on tribological performance? This would require reconfiguring the VLBT for elevated temperature.
- How would the HVPC coatings that showed excellent friction and wear behavior under lubricated VLBT test conditions, perform under more severe wear conditions of boundary lubrication and sooty oil?
- In the present Phase 2 work, the sliding partner was 8620 steel and the lubricant a commercial diesel engine oil. The effects of using other counterface materials and lubricants should be investigated.
- One primary Ti alloy was the focus of this work (Ti-6Al-4V). It is possible that another Ti would be more amenable to the surface treatments than this one.
- Only one variant of each surface treatment was investigated in this work, yet it could be argued that the performance of a given treatment could be optimized for specific applications and substrates. Given the leading candidates identified here, a follow-on effort to optimize their performance could be conducted.

5.0 Publications

In addition to this final report, the following publications were produced:

1) D. G. Bansal, O. L. Eryilmaz, and P. J. Blau (2011) "Surface engineering to improve the durability and lubricity of Ti-6Al-4V alloy," *Wear*, Vol. 271, pp. 2006-2015. Also presented at the 2011 International Conference on Wear of Materials, April 2011, Philadelphia, PA.

2) D. G. Bansal, M. Kirkham, and P. J. Blau (2013 expected) "Effects of combined diffusion treatments and cold working on the sliding friction and wear behavior of Ti-6Al-4V," accepted for presentation at the 2013 International Conference on Wear of Materials, and publication in a special issue of the journal *Wear*.

3) P. J. Blau, K. M. Cooley, D. Bansal, I. Smid, T. J. Eden, M. Neshastehriz, J. K. Potter, and A. E. Segall (2013 expected) “Spectrum loading effects on the running-in of lubricated bronze and surface-treated titanium against alloy steel,” accepted for presentation at the 2013 International Conference on Wear of Materials, and publication in a special issue of the journal *Wear*.

6.0 Commercialization Possibilities

Several of the surface engineering approaches described here are commercially available thanks to prior investment by the defense and aerospace industry, but they have not been considered for components in diesel engines because titanium parts are not yet commonplace in heavy vehicle propulsion systems. The issue is whether the relevant engineering and cost factors alone would warrant such an investment. There is no question that Ti-alloys are lighter than steel and cast iron and there is no doubt that they are far more corrosion resistant as well.

Certain high-end performance vehicles, like the Porche 911 GT3 already use titanium connecting rods, however, in racing applications there are some concerns about fatigue life [20].

Therefore, the surface engineering technology investigated here under simulative conditions, if proven to perform well, could promote renewed consideration of Ti-alloys for fuel efficient diesel engines. Barring the highly unlikely probability of regulatory mandate requiring the use of Ti in engines, the following four issues will control its future introduction:

- Cost of the finished component, including raw materials, machining, and finishing
- Demonstrated equal or better performance to existing components
- Ease of integration/substitution of Ti alloy connecting rods and other bearing surface-containing parts into the engine designs, including the cost of developing manufacturing and quality control specifications for those new parts
- Availability and performance of existing lubricants with Ti-containing bearing surfaces

One of the contributors to the cost of Ti alloys in aerospace components has been the wastage of material during conventional manufacturing and finishing; however, the landscape of manufacturing has been changing in recent years to include innovative additive manufacturing processes that may reduce the cost of Ti components. One aspect of performance, the friction and wear of contact surfaces, was addressed here, but the integration of Ti alloys into design includes other properties like fatigue life, thermal expansion, and thermal conductivity (see Appendix C). For example, if a piston swells relative to the cylinder liner as it heats, there is a danger of scuffing or seizure. Therefore, the thermal behavior of the moving part must be considered in relationship to that of the surrounding structures.

If durability and material compatibility issues can suitably be validated, fuel economy increases and ability to carry more load if the empty weight of a vehicle is reduced, then Ti could look better to the diesel engine community. Beyond that, friction and wear concerns for even the most promising candidate surface treatments will force designers to use inserts in a conservative, near-term approach in parts like connecting rod bearings.

7.0 Conclusions

The least risky course to further the use of lightweight, corrosion-resistant titanium alloy components as connecting rods, or other parts in which there is a need for repeated moving contact with another engine part, is to use inserts of conventional bearing materials rather than to attempt to treat or coat the surfaces of titanium itself.

Two types of friction and wear tests were used in this study of a variety of candidate coatings and surface treatments for Ti alloys. A ball-on-flat (point contact) reciprocating sliding test was used during the first phase of the effort to down-select candidates for the second phase. That test was hard on the materials, did not form a thick, stable lubricating film, and as a result produced relatively high friction coefficients typical of boundary lubrication or starved contact. The second type of test used unidirectional sliding with a cylinder-on-flat (line contact) which operated in low-friction regimes of lubrication in which the lubricant and surface finish seemed more relevant than the materials involved. In that regime, the materials did not make contact as often or with as much pressure. Not surprisingly, the candidate material choices ranked differently in the two test methods.

A satellite study of the effects of combined surface treatments, a mechanical treatment plus a diffusion treatment, showed little benefit under lubricant-starved test conditions, other than a temporary reduction in friction and wear for the first few meters of sliding. After that period, both treated surfaces showed the high friction and wear of the non-treated Ti alloy.

If the lubricant can be reliably supplied to maintain thick-film conditions during engine operation, then several of the Ti treatments and coatings could perform as well or better than a reference leaded bronze alloy. Two of the best candidates, based on the results of both types of screening tests, were a CrN commercial coating and a more experimental cold-sprayed Ni-B composite.

In summary, it would be prudent simply to avoid the use of titanium bearing surfaces entirely and use inserts of existing bearing materials in, for example, lightweight connecting rods. In less heavy-duty bearing surfaces, the results suggest limited future use of surface treated titanium for components that operate exclusively under mixed film or better, thick film lubrication conditions. Diffusion treatments like carburizing, nitriding, and oxygen diffusion treatments may provide limited protection for Ti, but are risky because once removed, the harsh friction and wear of titanium substrate occurs. A reasonable course could be to use a non-Ti, substrate durable hard coating such as CrN.

8.0 Cited References

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9.0 Symbols and Nomenclature

Symbols

h	film thickness of a lubricant
σ_l	root-mean-square surface roughness of component 1
σ_o	composite surface roughness
μ	friction coefficient (see also CoF)
HRC	Rockwell hardness number (C scale)
HV	Vickers microindentation hardness number
P	applied (normal) load (also designated as w in film thickness calculations)
V	wear volume

Nomenclature

60NiTi	An alloy developed by NASA containing 60 wt.% Ni and 40 wt. % Ti
Anod	Anodic surface filled with polymer used as a coating (see Tiodize™, Appendix A)
BB	(see Brnz)
Brnz	Bronze alloy CDA 932 (also referred to as ‘BB’ in some figures)
CR	Carburized surface treatment (also referred to as ‘Carb’)
Carb	(also referred to as “CR” in some figures and tables)
CoF	Coefficient of friction (see also μ)
CrN	Chromium nitride coating
CuNiIn	An electroplated coating of Cu, Ni, and In; often applied to connecting rod bearing inserts
DLC	Diamond coating (Note: This is not the usual designation for diamond coatings of this type but was used in this report for convenience in reporting.)
HVPC	High velocity powder consolidation coating
IR	Infrared melted composite surface layer containing Ti and TiB ₂ particles
NAMB	Nanocomposite of Al, Mg, and B developed at Ames Laboratory in DOE-funded program.
NR	Nitrided surface treatment
Λ -ratio	in lubrication theory, the ratio of the lubricant film thickness to the composite roughness of the two facing bearing surfaces
OD	Oxygen diffusion treated surface
ORNL	Oak Ridge National Laboratory
PL	Planished surface
RPOF	Reciprocating pin-on-flat wear test method
SAE	Society of Automotive Engineers (sometimes uses as a designation for materials and specifications)
SP	Shot peened surface
Ti64	Alloy Ti-6 wt.%Al-4 wt. % V
TiN	Titanium nitride coated surface
VLBT	Variable load bearing test rig

Appendix A. Materials and Sources – Phase 1 Tests

The slider material in Phase 1 studies was a Cr-containing bearing steel AISI 52100 in the form of finished 9.53 mm diameter, Grade 10 balls. The following materials and surface treatments were used for the $25.4 \times 25.4 \times 6.3$ mm thickness test coupons in this study. The designations in parentheses are used to identify the materials in subsequent discussions and data compilations. When multiple treatments are used, the designations are separated by a plus sign '+' in the order that the treatment was applied (i.e., treatment 1+ treatment 2).

Titanium alloy baseline material: Wrought, hot-rolled strip was purchased from a commercial source [A1], cut into test coupons without further heat treatment, and used for both substrate materials and for non-treated tests. The nominal composition of this material, according to ASTM B348-10, designated as Grade 5 titanium, in wt%, is: 5.5-6.75 Al, 3.5-4.5 V, 0.4 (max.) Fe, 0.20 (max.) O, 0.08 (max.) C, 0.05 (max.) N, 0.015 (max.) H, and balance Ti. The bulk hardness of the Ti64 specimens was 37.39 (± 0.43) HRC.

Bearing bronze baseline material: Alloy CDA 932 is a popular leaded bronze alloy used in plain bearings for engines. It was obtained from a commercial source [A2] as bar stock and has the following typical composition in wt%: 6.3-7.5 Sn, 6.0-8.0 Pb, 1.0-4.0 Zn, 0.35 Sb, 0.20 Fe, 1.0 Ni, 0.15 P, 0.08 S, 0.005 Si, 0.005 Al, and 83.0 (nominal) Cu. The alloy was tested in the as-received condition after surface finishing in successive steps to a 1.0 μm diamond lapped finish.

Nitinol: 60Ni-40Ti a NASA developed alloy was cast at Oak Ridge National Laboratory (ORNL). The bulk hardness of the as-cast alloy was 56.87 (± 1.4) HRC, the high variation in the hardness measurement is due to the uneven cast surface of the alloy. In order to machine the alloy to the required dimensions, the specimen was heat treated as per a process recommended by Julien [A3]. This treatment consists of heating the specimen in a furnace at 700 °C for 1 hour followed by gradual cooling in the furnace until the room temperature is reached. This reduces the bulk hardness to around 39.58 (± 0.58) HRC, and makes the alloy amenable to machining or grinding and this version of the alloy is designated as **60NiTi**. The thickness of the oxide layer formed as a result of this heat treatment was around 7.4 μm . The test coupon was then polished sequentially with 400-grit, 600-grit and 1 μm diamond paste in the same order. Julien also recommended a subsequent heat treatment to harden the surface with a black oxide. This heat treatment involved heating the specimen in a furnace up to 945 °C and holding it at that temperature for 15 minutes and then quenching it quickly in water. This covers the surface with a hard black colour oxide. The test coupon was then lightly polished with a 1 μm diamond paste. The bulk hardness increased to 41 (± 0.59) HRC and this version of the alloy is designated as **60NiTi + HT**. The thickness of the oxide layer formed as a result of this heat treatment was around 10 μm , resulting in a total oxide thickness of around 17–18 μm .

Oxygen-diffused Ti64 : The Ti64 base material was used as the substrate for an oxygen diffusion treatment consisting of exposure to 800 °C in air for two hours, followed by an air cool [A4]. Loose surface scales were lightly abraded away with 600 grit sand paper to produce a smooth surface with an enriched oxygen subsurface zone approximately 20 μm thick.

Nitrided Ti64: The nitriding of the Ti64 test coupons was performed by Solar Atmospheres [A5]. The furnace was pump down into hard vacuum (below 5×10^{-4} torr) to start the cycle with a very low oxygen atmosphere. The furnace was then slowly ramped to 800 °C ($+0^{\circ}/-14^{\circ}\text{C}$). This was followed with introduction of a partial pressure of nitrogen gas with a constant flow rate through the furnace, in order to maintain a dynamic atmosphere. The furnace was held at that temperature for two hours and quenched with a nitrogen gas backfill and internal furnace blower fan.

Carburized Ti64: The carburizing of the Ti64 test coupons was performed by Solar Atmospheres [A5]. The furnace vacuum and the ramp up were same as that for the nitriding process. The carburizing gas used was acetylene. A total of 21 boost/diffuse sequences were used for a total cycle time of seven hours. The boost was in a partial pressure of acetylene and the diffusion in a partial pressure of argon. The boosts become shorter in time and the diffusion longer as the cycle progresses. The furnace was cooled with an argon gas backfill and internal furnace blower fan at the end of the cycle.

Microcomposite by infrared surface processing: A novel, infrared surface treatment was used to produce Ti matrix composite coating deposits approximately 300 to 500 μm thick, containing micrometer-sized hard particles. They were prepared by rapid surface melting of mixed powders (Ti64 plus approximately 25% by volume of -100 mesh TiB_2 powder) using a Vortek™ plasma arc lamp. The lamp is a direct current, water-wall-stabilized, high-pressure argon arc lamp with a maximum power of 300 kW at an arc current of 1000 A. The heating was conducted in a water cooled box with flowing argon gas. The arc lamp was configured with a 35 cm long parabolic reflector to create line source heating with a half intensity width of about 1 cm. Powder coated samples were traversed at 6 to 7 mm/s in two passes, a preheat pass at 600 to 700 A, followed immediately by a melt pass at 875 to 900 A. Prior to IR treatment, powder blends were first mixed into an aqueous solution with a polymer binder. The suspended powder particles were then sprayed onto pre-weighed plates of Ti-6Al-4 V, dried in air for 12 hours, and heated to 650 °C to remove the remaining binder. The test specimen was exposed to 485 °C in air for approximately 3 hrs after polishing, and as a result had a thin layer of oxide.

Nanocomposite coating containing Al-Mg-B / TiB_2 : This coating material was developed by Ames Laboratory and Eaton Corporation under a research contract sponsored by the U.S. Department of Energy, Industrial Technologies Program. With a thickness of approximately 4–5 mm, it is produced by a multiple electrode chamber deposition process with a thin carbon layer [A6]. This coating was selected based on prior experience on a different research project on advanced coating materials.

Shot peening Ti64 : The test surfaces of the coupons were shot peened with the following parameters, 230H cast shot with 0.008-0.012" A (Almen intensity) and coverage of 125%. The same parameters were used for shot peening oxygen diffusion treated Ti64 coupons (OD + SP), nitrided Ti64 coupons (NR + SP) and carburized Ti64 coupons (CR + SP). Shot peening was performed by Metal Improvement Company, Milwaukee WI [A7].

TiN coating: TiN coating on Ti64 coupons was provided by Phygen Coatings Inc. [A8]. The coating thickness was in the range of 3–4 μm .

Chromium nitride coating: CrN or FortiPhy™ coating on Ti64 coupons was provided by Phygen Coatings Inc.[A8] using their patented physical vapor deposition process. The coating thickness was around 3 μm .

Diamond films: The nearly frictionless diamond-like-carbon coatings, provided by Argonne National Laboratory (ANL), were grown from pure CH_4 by using a plasma enhanced chemical vapor depositions (PE-CVD) technique [A9-A11]. The coating thickness was in the range of 1–1.4 μm .

Cu-Ni-In coating: The bearing bronze and Ti64 test coupons were coated with Cu-Ni-In using a thermal spray process. The coating was performed by Ellison Surface Technologies [A12] using their proprietary composition and process parameters. The coating thickness was in the range of 30 – 34 μm .

Anodizing: TIODIZE, an anodizing process for titanium, was performed by TIODIZE Technologies [A13]. The TIODIZE IV process involves impregnating TIOLON X40 PTFE into the surface to provide

low friction and anti-galling for longer life at higher loads. This process was selected from an internet search of coating processes for titanium. According to TIODIZE Technologies, this process prevents the failure due to galling and reduces the coefficient of friction.

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- [A13] TIODIZE Technologies, Huntington Beach, CA

Appendix B. Table of VLBT Data for Cycle 10 of Spectrum Tests

VLBT#	25				VLBT#	28				VLBT#	29				VLBT#	30			
SURFACE	Bare CDA 932 Bronze				SURFACE	Bare CDA 932 Bronze				SURFACE	Non-treated Ti-6Al-4V				SURFACE	Non-treated Ti-6Al-4V			
		Ave COF	0.019				Ave COF	0.016				Ave COF	0.234				Ave COF	0.238	
		StdDev	0.001				StdDev	0.002				StdDev	0.042				StdDev	0.043	
		Max COF	0.020				Max COF	0.019				Max COF	0.300				Max COF	0.307	
		Min COF	0.017				Min COF	0.014				Min COF	0.115				Min COF	0.123	
Time	Load, P	1/P	Fric Coeff		Time	Load, P	1/P	Fric Coeff		Time	Load, P	1/P	Fric Coeff		Time	Load, P	1/P	Fric Coeff	
(s)	(N)	(N ⁻¹)	(n/d)		(s)	(N)	(N ⁻¹)	(n/d)		(s)	(N)	(N ⁻¹)	(n/d)		(s)	(N)	(N ⁻¹)	(n/d)	
1082	31.07	0.032185	0.02		1080.017	54.31507	0.018411	0.016675		1086	29.67	0.033704	0.188		1082.022	33.72528	0.029651	0.241128	
1084	30.98	0.032279	0.019		1082.019	35.00508	0.028567	0.014563		1088	29.81	0.033546	0.176		1084.023	29.67837	0.033695	0.23013	
1086	31.12	0.032134	0.019		1084.017	30.71728	0.032555	0.014247		1090	29.71	0.033659	0.158		1086.024	29.71324	0.033655	0.208636	
1088	30.96	0.0323	0.019		1086.018	30.67299	0.032602	0.014439		1092	29.78	0.03358	0.153		1088.023	29.66233	0.033713	0.175767	
1090	30.95	0.03231	0.019		1088.019	30.78417	0.032484	0.013658		1094	29.89	0.033456	0.148		1090.024	29.762	0.0336	0.160862	
1092	31	0.032258	0.019		1090.019	30.90003	0.032362	0.014637		1096	29.74	0.033625	0.143		1092.015	29.71663	0.033651	0.157116	
1094	30.97	0.032289	0.02		1092.012	30.95183	0.032308	0.013926		1098	29.76	0.033602	0.125		1094.025	29.59632	0.033788	0.146446	
1096	31.07	0.032185	0.018		1094.018	30.86226	0.032402	0.014043		1100	29.75	0.033613	0.115		1096.023	29.66279	0.033712	0.141697	
1098	31.07	0.032185	0.019		1096.017	30.84146	0.032424	0.015219		1102	49.59	0.020165	0.168		1098.023	29.62941	0.03375	0.140945	
1100	50.75	0.019704	0.018		1098.018	30.91007	0.032352	0.014267		1104	53.74	0.018608	0.191		1100.022	29.71996	0.033647	0.123311	
1102	54.74	0.018268	0.018		1100.018	30.83201	0.032434	0.01425		1106	53.8	0.018587	0.195		1102.021	49.24367	0.020307	0.169281	
1104	54.75	0.018265	0.018		1102.018	50.32323	0.019872	0.016469		1108	53.52	0.018685	0.211		1104.025	53.39136	0.01873	0.192065	
1106	54.71	0.018278	0.018		1104.018	54.77267	0.018257	0.016875		1110	53.74	0.018608	0.211		1106.023	53.51514	0.018686	0.200783	
1108	54.25	0.018433	0.018		1106.018	54.69685	0.018283	0.016213		1112	71.25	0.014035	0.214		1108.023	53.40669	0.018724	0.204983	
1110	72.38	0.013816	0.019		1108.018	54.57456	0.018324	0.016918		1114	74.62	0.013401	0.235		1110.022	53.42701	0.018717	0.209747	
1112	75.91	0.013173	0.019		1110.018	54.71688	0.018276	0.017392		1116	74.62	0.013401	0.249		1112.023	71.10868	0.014063	0.22691	
1114	75.98	0.013161	0.019		1112.018	72.13467	0.013863	0.017949		1118	74.6	0.013405	0.248		1114.026	74.42363	0.013437	0.247952	
1116	76.02	0.013154	0.019		1114.019	75.81154	0.013191	0.018437		1120	74.62	0.013401	0.252		1116.023	74.53645	0.013416	0.260201	
1118	75.95	0.013167	0.019		1116.018	75.68039	0.013213	0.018365		1122	49.53	0.02019	0.266		1118.022	74.39841	0.013441	0.26265	
1120	50.85	0.019666	0.018		1118.019	75.66412	0.013216	0.018012		1124	43.4	0.023041	0.264		1120.022	74.39132	0.013442	0.272374	
1122	44.64	0.022401	0.019		1120.018	75.73618	0.013204	0.018048		1126	43.49	0.022994	0.251		1122.024	49.31407	0.020278	0.30647	
1124	44.69	0.022376	0.019		1120.018	75.73618	0.013204	0.018048		1128	43.51	0.022983	0.233		1124.023	43.2914	0.023099	0.26574	
1126	44.51	0.022467	0.019		1122.019	50.54711	0.019784	0.016312		1130	43.53	0.022973	0.227		1126.024	43.33505	0.023076	0.238486	
1128	44.74	0.022351	0.02		1124.017	44.51286	0.022465	0.015579		1132	43.4	0.023041	0.221		1128.023	43.18522	0.023156	0.218017	
1130	44.73	0.022356	0.019		1126.019	44.67845	0.022382	0.015067		1134	43.31	0.023089	0.208		1130.023	43.10038	0.023202	0.220202	
1132	44.73	0.022356	0.018		1128.017	44.57591	0.022434	0.015411		1136	51.72	0.019335	0.205		1132.024	43.28695	0.023102	0.217108	
1134	53.04	0.018854	0.019		1130.017	44.51191	0.022466	0.015128		1138	69.29	0.014432	0.237		1134.024	43.26354	0.023114	0.212379	
1136	70.57	0.01417	0.019		1132.018	44.51039	0.022467	0.015461		1140	69.34	0.014422	0.248		1136.023	50.97638	0.019617	0.206354	
1138	70.62	0.01416	0.019		1134.02	44.45712	0.022494	0.015687		1142	69.41	0.014407	0.253		1138.023	69.03874	0.014485	0.234684	
1140	70.69	0.014146	0.019		1136.02	53.16291	0.01881	0.016788		1144	69.45	0.014399	0.257		1140.023	69.03461	0.014485	0.249543	
1142	70.46	0.014192	0.019		1138.019	70.13963	0.014257	0.017915		1146	78.15	0.012796	0.239		1142.024	69.15058	0.014461	0.254846	
1144	80.1	0.012484	0.018		1140.018	70.39456	0.014206	0.017666		1148	96.57	0.010355	0.235		1144.044	69.12107	0.014467	0.259577	
1146	101.83	0.00982	0.02		1142.021	70.38704	0.014207	0.01752		1150	96.52	0.010361	0.249		1146.023	78.09868	0.012804	0.255782	
1148	101.68	0.009835	0.019		1144.018	70.27417	0.01423	0.01796		1152	96.51	0.010362	0.251		1148.023	97.00654	0.010309	0.247731	
1150	101.43	0.009859	0.019		1146.017	80.74118	0.012385	0.018491		1154	96.56	0.010356	0.256		1150.024	96.744	0.010337	0.253429	
1152	101.76	0.009827	0.019		1148.018	102.7083	0.009736	0.018916		1156	87.26	0.01146	0.274		1152.022	97.168	0.010291	0.26126	
1154	90.89	0.011002	0.019		1150.017	101.9902	0.009805	0.018632		1158	69.19	0.014453	0.3		1154.023	97.00852	0.010308	0.263862	
1156	70.13	0.014259	0.018		1152.018	102.0557	0.009799	0.018586		1160	69.19	0.014453	0.285		1156.025	87.5245	0.011425	0.281239	
1158	70.37	0.014211	0.019		1154.02	101.6316	0.009839	0.018348		1162	69.02	0.014489	0.279		1158.022	68.80226	0.014534	0.306571	
1160	70.39	0.014207	0.018		1156.02	90.97183	0.010992	0.018214		1164	69.01	0.014491	0.281		1160.024	68.83314	0.014528	0.305386	
1162	70.44	0.014196	0.018		1158.018	70.18256	0.014249	0.016488		1166	60.87	0.016428	0.291		1162.024	68.74274	0.014547	0.294789	
1164	61.9	0.016155	0.019		1160.018	70.0428	0.014277	0.016446		1168	43.31	0.023089	0.287		1164.026	68.92337	0.014509	0.284747	
1166	44.59	0.022427	0.018		1162.017	69.89147	0.014308	0.017008		1170	43.28	0.023105	0.283		1166.023	60.51005	0.016526	0.283091	
1168	44.59	0.022427	0.018		1164.019	70.08349	0.014269	0.016924		1172	43.33	0.023079	0.267		1168.026	43.1236	0.023189	0.289872	
1170	44.63	0.022406	0.018		1166.018	61.69962	0.016208	0.016316		1174	43.26	0.023116	0.265		1170.025	43.11212	0.023195	0.275513	
1172	44.65	0.022396	0.018		1168.017	44.23261	0.022608	0.014505		1176	43.22	0.023137	0.253		1172.024	43.09533	0.023204	0.257795	
1174	44.59	0.022427	0.018		1170.018	44.24054	0.022604	0.014812		1178	43.27	0.023111	0.245		1174.025	43.098	0.023203	0.239938	
1176	44.62	0.022411	0.018		1172.018	44.31245	0.022567	0.014787		1180	43.2	0.023148	0.247		1176.025	43.19205	0.023152	0.225532	

VLBT#	35				VLBT#	36				VLBT#	39				VLBT#	40			
SURFACE	OD - Ti-6Al-4V				SURFACE	OD - Ti-6Al-4V				SURFACE	Nitrided Ti-6Al-4V				SURFACE	Nitrided Ti-6Al-4V			
	Ave COF			0.021		Ave COF			0.040		Ave COF			0.029		Ave COF			0.025
	StdDev			0.006		StdDev			0.004		StdDev			0.013		StdDev			0.010
	Max COF			0.035		Max COF			0.052		Max COF			0.066		Max COF			0.048
	Min COF			0.013		Min COF			0.033		Min COF			0.015		Min COF			0.011
Time	Load, P	1/P	Fric Coeff		Time	Load, P	1/P	Fric Coeff		Time	Load, P	1/P	Fric Coeff		Time	Load, P	1/P	Fric Coeff	
(s)	(N)	(N^-1)	(n/d)		(s)	(N)	(N^-1)	(n/d)		(s)	(N)	(N^-1)	(n/d)		(s)	(N)	(N^-1)	(n/d)	
1082.022	32.68128	0.030599	0.014817		1086.023	32.14721	0.031107	0.034859		1082.022	36.44802	0.027436	0.015754		1082.014	32.33005	0.030931	0.011164	
1082.022	32.68128	0.030599	0.014817		1088.023	32.08427	0.031168	0.036507		1084.021	36.35966	0.027503	0.015302		1084.015	32.37649	0.030887	0.01142	
1084.023	32.47454	0.030793	0.014497		1090.024	32.07922	0.031173	0.035998		1086.023	36.43872	0.027443	0.015155		1086.015	32.31832	0.030942	0.011533	
1086.022	32.46952	0.030798	0.014418		1094.038	31.95783	0.031291	0.036328		1088.023	36.38382	0.027485	0.015356		1088.015	32.39798	0.030866	0.012029	
1088.023	32.54115	0.03073	0.014987		1096.024	32.11403	0.031139	0.0355		1090.008	36.37335	0.027493	0.015681		1090.006	32.33031	0.030931	0.011175	
1090.013	32.52654	0.030744	0.014542		1098.023	32.15166	0.031103	0.036412		1092.034	36.34672	0.027513	0.015569		1092.016	32.14839	0.031106	0.011041	
1092.022	32.4583	0.030809	0.014859		1100.023	32.05064	0.031201	0.036416		1094.021	36.47741	0.027414	0.015029		1094.015	32.42307	0.030842	0.01242	
1094.024	32.6397	0.030638	0.014211		1102.024	51.54025	0.019402	0.039536		1096.02	36.49581	0.0274	0.015924		1096.016	32.45954	0.030808	0.011262	
1096.023	32.62142	0.030655	0.013488		1104.023	55.95392	0.017872	0.040217		1098.023	36.35564	0.027506	0.015181		1098.017	32.39847	0.030866	0.011275	
1098.023	32.65471	0.030623	0.013661		1106.023	55.98188	0.017863	0.039862		1100.021	57.11726	0.017508	0.026829		1100.015	52.60001	0.019011	0.023093	
1100.023	52.27656	0.019129	0.018083		1108.024	56.09561	0.017827	0.039814		1102.021	61.38394	0.016291	0.027465		1102.014	57.52214	0.017385	0.025562	
1102.022	56.49956	0.017699	0.017648		1110.025	56.05704	0.017839	0.040179		1104.041	61.44756	0.016274	0.027992		1104.016	57.14377	0.0175	0.025557	
1104.021	56.62288	0.017661	0.018243		1112.023	73.4872	0.013608	0.04143		1106.023	61.24429	0.016328	0.027707		1106.016	57.29789	0.017453	0.025382	
1106.023	56.61843	0.017662	0.018724		1114.024	77.19759	0.012954	0.042099		1108.023	61.45531	0.016272	0.02768		1108.014	57.22657	0.017474	0.026048	
1108.023	56.73971	0.017624	0.017771		1116.024	77.04603	0.012979	0.04301		1110.022	78.32798	0.012767	0.033451		1110.015	74.66336	0.013393	0.03074	
1110.023	74.29288	0.01346	0.024666		1118.023	77.20376	0.012953	0.042252		1112.021	81.8064	0.012224	0.033913		1112.015	77.75561	0.012861	0.031822	
1112.023	77.65673	0.012877	0.025381		1120.023	76.99791	0.012987	0.043072		1114.021	81.89651	0.012211	0.034305		1114.015	77.85881	0.012844	0.031634	
1114.024	77.78619	0.012856	0.025531		1122.023	51.52668	0.019407	0.040741		1116.02	81.8481	0.012218	0.034268		1116.014	77.7369	0.012864	0.031689	
1116.023	77.8008	0.012853	0.025766		1124.023	45.75903	0.021854	0.039031		1118.021	81.94134	0.012204	0.033992		1118.016	77.80977	0.012852	0.031876	
1118.023	77.69622	0.012871	0.024861		1126.022	45.59728	0.021931	0.03763		1120.023	56.64511	0.017654	0.025053		1120.015	52.37539	0.019093	0.022932	
1120.022	52.47568	0.019056	0.018855		1126.022	45.59728	0.021931	0.03763		1122.022	50.42936	0.01983	0.021739		1122.015	46.37376	0.021564	0.020036	
1122.022	46.54889	0.021483	0.01705		1128.026	45.64119	0.02191	0.039297		1124.032	50.49663	0.019803	0.022264		1124.015	46.30936	0.021594	0.019497	
1124.023	46.41355	0.021545	0.016998		1130.023	45.83141	0.021819	0.038216		1126.022	50.45146	0.019821	0.022219		1126.015	46.29619	0.0216	0.019506	
1126.02	46.33733	0.021581	0.016917		1132.023	45.6391	0.021911	0.037145		1128.022	50.36539	0.019855	0.022041		1128.015	46.16213	0.021663	0.020099	
1128.023	46.44139	0.021533	0.01586		1134.024	45.69093	0.021886	0.038241		1130.038	50.49296	0.019805	0.022315		1130.015	46.2645	0.021615	0.019711	
1130.023	46.36267	0.021569	0.016879		1136.024	53.92846	0.018543	0.038424		1132.022	50.39719	0.019842	0.021956		1132.015	46.17531	0.021657	0.019354	
1132.024	46.23218	0.02163	0.017238		1138.023	71.75753	0.013936	0.040652		1134.022	59.19848	0.016892	0.026929		1134.015	54.75978	0.018262	0.02463	
1134.024	54.70031	0.018281	0.019481		1140.024	71.71107	0.013945	0.040821		1136.023	76.75406	0.013029	0.032418		1136.015	72.68907	0.013757	0.030838	
1136.022	72.4112	0.01381	0.023699		1142.023	71.74296	0.013939	0.040669		1138.021	76.69695	0.013038	0.032581		1138.017	72.45182	0.013802	0.030823	
1138.023	72.24518	0.013842	0.023135		1144.037	71.70364	0.013946	0.04072		1140.022	76.56066	0.013062	0.032603		1140.017	72.40205	0.013812	0.030609	
1140.023	72.41711	0.013809	0.023697		1146.026	80.85754	0.012367	0.044614		1142.022	76.66105	0.013044	0.032386		1142.017	72.46221	0.0138	0.030543	
1142.022	72.33552	0.013824	0.023301		1148.023	103.3555	0.009675	0.051145		1144.022	106.1533	0.00942	0.046761		1144.016	89.88111	0.011126	0.036328	
1144.023	82.44334	0.01213	0.027468		1150.025	102.6997	0.009737	0.05174		1146.023	188.0089	0.005319	0.063907		1146.015	154.0014	0.006493	0.047549	
1146.025	107.8596	0.009271	0.035211		1152.024	103.1217	0.009697	0.051539		1148.023	188.0089	0.005319	0.064822		1148.015	153.927	0.006497	0.047977	
1148.024	106.296	0.009408	0.034685		1154.024	103.0465	0.009704	0.051488		1150.021	188.0089	0.005319	0.065896		1150.014	153.8767	0.006499	0.048064	
1150.022	105.3917	0.009488	0.034667		1156.022	92.48605	0.010812	0.048269		1152.022	188.0089	0.005319	0.066199		1152.015	150.6519	0.006638	0.047898	
1152.025	104.4584	0.009573	0.03466		1158.023	71.64337	0.013958	0.040964		1154.022	143.9854	0.006945	0.060215		1154.014	117.0945	0.00854	0.042759	
1154.023	94.762	0.010553	0.032589		1160.024	71.45651	0.013995	0.041207		1156.024	75.71861	0.013207	0.032694		1156.038	70.97123	0.01409	0.029455	
1156.024	72.24147	0.013842	0.024112		1162.024	71.57768	0.013971	0.040373		1158.035	75.86101	0.013182	0.032748		1158.016	70.74485	0.014135	0.029594	
1158.023	72.25356	0.01384	0.023778		1164.023	71.53871	0.013978	0.039324		1160.022	75.67465	0.013214	0.032755		1160.017	71.18638	0.014048	0.029154	
1160.022	72.18738	0.013853	0.023614		1166.023	71.53871	0.013978	0.039324		1162.023	75.79131	0.013194	0.032845		1162.017	71.04335	0.014076	0.029682	
1162.023	72.30438	0.01383	0.023887		1166.035	63.24458	0.015812	0.039592		1164.032	67.59045	0.014795	0.030611		1164.015	62.97906	0.015878	0.026994	
1164.024	63.94046	0.01564	0.022078		1168.024	45.46645	0.021994	0.038744		1166.023	50.55833	0.019779	0.022732		1166.015	46.157	0.021665	0.020242	
1166.024	46.20296	0.021644	0.017192		1170.024	45.62911	0.021916	0.037228		1168.024	50.59056	0.019767	0.023027		1168.016	46.09598	0.021694	0.02061	
1168.024	46.28323	0.021606	0.015962		1172.023	45.53466	0.021961	0.035719		1170.037	50.76611	0.019698	0.023559		1170.016	46.22656	0.021633	0.020001	

VLBT#	41				VLBT#	42				VLBT#	43				VLBT#	44			
SURFACE	TiN coated Ti-6Al-4V				SURFACE	TiN coated Ti-6Al-4V				SURFACE	CrN (Fortiphy TM)				SURFACE	CrN (Fortiphy TM)			
	Ave COF		0.012			Ave COF		0.016			Ave COF		0.014			Ave COF		0.011	
	StdDev		0.005			StdDev		0.004			StdDev		0.006			StdDev		0.003	
	Max COF		0.027			Max COF		0.029			Max COF		0.035			Max COF		0.020	
	Min COF		0.008			Min COF		0.011			Min COF		0.009			Min COF		0.008	
Time	Load, P	1/P	Fric Coeff		Time	Load, P	1/P	Fric Coeff		Time	Load, P	1/P	Fric Coeff		Time	Load, P	1/P	Fric Coeff	
(s)	(N)	(N ⁻¹)	(n/d)		(s)	(N)	(N ⁻¹)	(n/d)		(s)	(N)	(N ⁻¹)	(n/d)		(s)	(N)	(N ⁻¹)	(n/d)	
1080.024	36.02016	0.027762	0.008743		1082.024	31.93208	0.031316	0.010519		1080.023	42.59227	0.023478	0.011505		1082.023	34.16436	0.02927	0.010985	
1082.023	31.76645	0.03148	0.008732		1084.023	31.94565	0.031303	0.011091		1082.022	38.62176	0.025892	0.01221		1084.022	34.20026	0.02924	0.010282	
1084.022	31.82689	0.03142	0.008588		1086.024	31.90415	0.031344	0.011575		1084.02	38.32404	0.026093	0.011732		1086.021	34.13336	0.029297	0.010585	
1086.022	32.10108	0.031152	0.009218		1088.024	31.86693	0.03138	0.011001		1086.023	38.36674	0.026064	0.011875		1088.022	34.1616	0.029273	0.010459	
1088.022	31.96008	0.031289	0.008725		1090.02	31.88071	0.031367	0.011924		1088.023	38.39444	0.026045	0.012365		1090.011	34.17948	0.029257	0.011137	
1090.014	31.93362	0.031315	0.008287		1092.023	32.16462	0.03109	0.011158		1090.013	38.41378	0.026032	0.012135		1092.023	34.13835	0.029293	0.01063	
1092.024	32.19685	0.031059	0.008628		1094.023	31.97828	0.031271	0.011567		1092.022	38.36293	0.026067	0.012348		1094.023	34.29522	0.029159	0.011142	
1094.024	31.92082	0.031328	0.008536		1096.023	31.86753	0.03138	0.011044		1094.023	38.38184	0.026054	0.012781		1096.022	34.1969	0.029242	0.01037	
1096.025	31.95589	0.031293	0.008178		1098.025	31.97412	0.031275	0.011464		1096.023	38.46897	0.025995	0.012123		1098.023	34.27327	0.029177	0.010973	
1098.022	32.08059	0.031172	0.009008		1100.024	52.20536	0.019155	0.015489		1098.021	38.55581	0.025936	0.01201		1100.021	54.84521	0.018233	0.010622	
1100.024	52.52018	0.01904	0.009594		1102.024	56.82966	0.017596	0.016248		1100.021	59.01718	0.016944	0.012086		1102.021	59.17799	0.016898	0.010085	
1102.024	56.89236	0.017577	0.010249		1104.024	56.9491	0.01756	0.015499		1102.021	63.06626	0.015856	0.012487		1104.021	59.23986	0.016881	0.00986	
1104.024	57.02362	0.017537	0.010022		1106.024	56.94718	0.01756	0.016179		1104.021	63.20614	0.015821	0.012879		1106.023	59.13772	0.01691	0.010068	
1106.024	56.96161	0.017556	0.009992		1108.024	56.81975	0.0176	0.015878		1106.024	63.24973	0.01581	0.011619		1108.021	59.03254	0.01694	0.010096	
1108.023	56.86799	0.017585	0.010304		1110.025	74.01861	0.01351	0.019107		1108.022	63.04933	0.015861	0.011851		1110.022	76.54396	0.013064	0.012564	
1110.024	74.21327	0.013475	0.014329		1112.024	77.30825	0.012935	0.019717		1110.024	80.33441	0.012448	0.013851		1112.022	79.77582	0.012535	0.012774	
1112.024	77.33381	0.012931	0.015147		1114.024	77.30759	0.012935	0.019264		1112.023	83.94039	0.011913	0.014253		1114.023	79.84602	0.012524	0.012368	
1114.024	77.46884	0.012908	0.014978		1116.024	77.39446	0.012921	0.019609		1114.022	83.80482	0.011932	0.014402		1116.021	79.68898	0.012549	0.012933	
1116.023	77.33622	0.012931	0.014998		1118.024	77.4334	0.012914	0.019006		1116.021	83.73941	0.011942	0.014376		1118.022	79.59829	0.012563	0.012488	
1118.024	77.44496	0.012912	0.015051		1120.024	51.77379	0.019315	0.015058		1118.022	83.95021	0.011912	0.014335		1120.021	54.45885	0.018362	0.009174	
1120.023	51.90414	0.019266	0.010251		1122.023	45.77079	0.021848	0.014614		1122.022	52.5116	0.019043	0.009655		1122.021	48.35101	0.020682	0.008649	
1122.025	45.91235	0.021781	0.008594		1124.025	45.91483	0.021779	0.013783		1124.022	52.5131	0.019043	0.009976		1124.022	48.16891	0.020706	0.00895	
1124.023	46.06451	0.021709	0.007897		1126.025	45.7098	0.021877	0.013633		1124.022	52.5131	0.019043	0.009976		1126.022	48.163	0.020763	0.009174	
1126.024	45.93012	0.021772	0.007751		1128.025	45.75532	0.021855	0.013707		1126.022	52.46635	0.01906	0.010114		1128.022	48.21784	0.020739	0.008473	
1128.025	45.84454	0.021813	0.008452		1130.025	45.98594	0.021746	0.013704		1128.024	52.38923	0.019088	0.008883		1130.022	48.22422	0.020736	0.009746	
1130.024	44.62562	0.022409	0.012558		1132.024	45.90616	0.021784	0.013692		1130.022	52.42221	0.019076	0.008837		1132.022	48.17904	0.020756	0.009322	
1132.024	44.43664	0.022504	0.013308		1134.023	54.38881	0.018386	0.015497		1132.022	52.48615	0.019053	0.010132		1134.023	56.67894	0.017643	0.01013	
1134.022	53.65042	0.018639	0.011446		1136.023	72.142	0.013862	0.018449		1134.022	60.78364	0.016452	0.01087		1136.021	74.30213	0.013459	0.012373	
1136.024	71.38246	0.014009	0.012348		1138.023	72.16706	0.013857	0.018179		1136.021	78.72036	0.012703	0.01366		1138.022	74.23153	0.013471	0.011567	
1138.023	71.16042	0.014053	0.011616		1140.024	72.05826	0.013878	0.018597		1138.022	78.73927	0.0127	0.013701		1140.023	74.1719	0.013482	0.012019	
1140.024	70.9399	0.014096	0.011139		1142.024	72.14381	0.013861	0.018334		1140.022	78.71893	0.012703	0.013513		1142.021	74.29272	0.01346	0.011613	
1142.024	71.64299	0.013958	0.012933		1144.025	89.55549	0.011166	0.022013		1142.023	78.67264	0.012711	0.013527		1144.023	99.85861	0.010014	0.015472	
1144.023	90.50692	0.011049	0.017755		1146.025	141.0082	0.007092	0.028721		1144.022	109.635	0.009121	0.023631		1146.023	172.3281	0.005803	0.020211	
1146.023	141.5645	0.007064	0.025889		1148.024	138.167	0.007238	0.028217		1146.023	187.8143	0.005324	0.034976		1148.022	171.3645	0.005836	0.01935	
1148.026	143.9132	0.006949	0.026433		1150.024	139.9343	0.007146	0.028638		1148.022	187.8143	0.005324	0.034433		1150.022	172.5478	0.005795	0.01955	
1150.023	141.4662	0.007069	0.026602		1152.024	136.9723	0.007301	0.028432		1150.023	187.8143	0.005324	0.034342		1152.022	173.8268	0.005753	0.019369	
1152.025	142.98	0.006994	0.026681		1154.023	110.1662	0.009077	0.025145		1152.023	187.8143	0.005324	0.034894		1154.024	129.5349	0.00772	0.016995	
1154.025	112.872	0.00886	0.023152		1156.024	70.70409	0.014143	0.017117		1154.022	145.2362	0.006885	0.028952		1156.022	73.20688	0.01366	0.010521	
1156.023	70.7085	0.014143	0.013099		1158.024	70.81816	0.014121	0.017219		1156.022	77.97371	0.012825	0.012844		1158.023	73.11668	0.013677	0.010968	
1158.024	70.69088	0.014146	0.012954		1160.025	70.56768	0.014171	0.017639		1158.023	77.91691	0.012834	0.012548		1160.023	73.10698	0.013679	0.010402	
1160.023	70.79296	0.014126	0.013173		1162.025	70.61019	0.014162	0.017299		1160.022	77.77503	0.012858	0.012969		1162.023	73.28924	0.013645	0.010468	
1162.024	70.82155	0.01412	0.012926		1164.024	62.70165	0.015949	0.01676		1162.023	77.96662	0.012826	0.012589		1164.022	65.26307	0.015323	0.010002	
1164.023	62.82746	0.015917	0.011806		1166.023	45.70926	0.021877	0.014066		1162.022	69.97867	0.01429	0.011785		1166.021	48.44502	0.020642	0.008811	
1166.024	45.57196	0.021943	0.008277		1168.025	45.69198	0.021886	0.013158		1166.022	52.4117	0.01908	0.010316		1168.024	48.19948	0.020747	0.008227	
1168.023	45.86609	0.021803	0.007881		1170.025	45.73503	0.021865	0.014012		1168.023	52.59517	0.019013	0.009612		1170.023	48.33789	0.020688	0.008585	
1170.023	45.68644	0.021888	0.008106		1172.025	45.77076	0.021848	0.013861		1170.022	52.65762	0.018991	0.009068		1172.024	48.29421	0.020706	0.008431	
1172.024	45.5298	0.021964	0.008202		1174.026	45.78258	0.021842	0.013641		1172.023	52.70056	0.018975	0.010391		1174.022	48.15146	0.020768	0.00917	
1174.024	45.88848	0.021792	0.008441		1176.025	45.67445	0.021894	0.013773		1174.023	52.5228	0.019039	0.009758		1176.023	48.24218	0.020729	0.008697	
1176.023	45.86153	0.021805	0.007784		1178.026	45.73087	0.021867	0.013835		1176.024	52.52719	0.019038	0.010325		1178.023	48.29215	0.020707	0.00	

VLBT#	45				VLBT#	46				VLBT#	47				VLBT#	48			
SURFACE	Carburized Ti-6Al-4V				SURFACE	Carburized Ti-6Al-4V				SURFACE	RMFC-4Z (Tiodize)				SURFACE	RMFC-4Z (Tiodize)			
	Ave COF		0.011			Ave COF		0.012			Ave COF		0.030			Ave COF		0.026	
	StdDev		0.002			StdDev		0.003			StdDev		0.001			StdDev		0.003	
	Max COF		0.017			Max COF		0.021			Max COF		0.033			Max COF		0.032	
	Min COF		0.009			Min COF		0.009			Min COF		0.028			Min COF		0.022	
Time	Load, P	1/P	Fric Coeff		Time	Load, P	1/P	Fric Coeff		Time	Load, P	1/P	Fric Coeff		Time	Load, P	1/P	Fric Coeff	
(s)	(N)	(N ⁻¹)	(n/d)		(s)	(N)	(N ⁻¹)	(n/d)		(s)	(N)	(N ⁻¹)	(n/d)		(s)	(N)	(N ⁻¹)	(n/d)	
1080.015	55.26753	0.018094	0.009645		1080.023	55.24495	0.018101	0.010084		1082.018	31.11626	0.032138	0.030523		1082.02	30.96007	0.0323	0.023547	
1082.016	35.88431	0.027867	0.010134		1082.021	35.98904	0.027786	0.010009		1084.019	31.16025	0.032092	0.032941		1084.02	31.05096	0.032205	0.022549	
1084.015	31.80204	0.031445	0.010049		1084.021	31.96724	0.031282	0.01039		1086.02	31.13778	0.032115	0.032585		1086.019	30.98969	0.032269	0.022741	
1086.016	31.80758	0.031439	0.011423		1086.023	31.84923	0.031398	0.010287		1088.018	31.21071	0.03204	0.031231		1088.019	30.85211	0.032413	0.023777	
1088.014	31.88125	0.031366	0.010317		1088.023	31.86826	0.031379	0.010419		1090.019	31.12191	0.032132	0.032808		1090.01	30.91562	0.032346	0.023225	
1090.016	31.991	0.031259	0.010837		1090.03	32.02627	0.031224	0.008652		1092.004	31.13704	0.032116	0.031837		1092.019	31.14389	0.032109	0.023025	
1092.007	31.9008	0.031347	0.011294		1092.012	32.01626	0.031234	0.010931		1094.018	31.23146	0.032019	0.032		1094.018	31.00567	0.032252	0.02382	
1094.031	31.89904	0.031349	0.011246		1094.022	32.09865	0.031154	0.009642		1096.018	31.15962	0.032093	0.032253		1096.018	31.20958	0.032041	0.022232	
1096.016	31.88762	0.03136	0.010819		1096.022	31.89058	0.031357	0.009577		1098.018	31.13948	0.032114	0.031735		1098.017	31.1095	0.032145	0.023936	
1098.015	31.91015	0.031338	0.010326		1098.022	31.89564	0.031352	0.010224		1100.03	31.16023	0.032092	0.032104		1100.017	51.49575	0.019419	0.026316	
1100.016	31.87273	0.031375	0.010914		1100.038	32.06421	0.031187	0.010493		1102.019	51.65633	0.019359	0.030115		1102.018	55.80401	0.01792	0.025608	
1102.015	52.56435	0.019024	0.010195		1102.023	52.64893	0.018994	0.011655		1104.019	55.89402	0.017891	0.030928		1104.018	55.96363	0.017869	0.026277	
1104.016	56.87227	0.017583	0.010589		1104.021	56.87868	0.017581	0.012185		1106.018	55.9006	0.017889	0.029621		1106.02	55.91688	0.017884	0.025955	
1106.017	56.77209	0.017614	0.010675		1106.022	56.75516	0.01762	0.011578		1108.019	55.94562	0.017874	0.030365		1108.019	55.95804	0.017871	0.025718	
1108.028	56.86788	0.017585	0.010358		1108.022	56.85997	0.017587	0.011173		1110.018	55.84039	0.017908	0.030031		1110.018	73.13388	0.013674	0.028198	
1110.014	57.07721	0.01752	0.010121		1110.022	56.89547	0.017576	0.011507		1112.018	73.25224	0.013651	0.030782		1112.019	76.48642	0.013074	0.02813	
1112.041	74.24191	0.013469	0.012088		1112.022	74.45188	0.013431	0.01373		1114.019	76.50411	0.013071	0.030538		1114.02	76.60896	0.013053	0.028619	
1114.016	77.45375	0.012911	0.012473		1114.023	77.44853	0.012912	0.013857		1116.02	76.61767	0.013052	0.030631		1116.02	76.33515	0.0131	0.028061	
1116.015	77.34923	0.012928	0.012595		1116.022	77.51792	0.0129	0.014137		1118.019	76.63911	0.013048	0.030622		1118.018	76.38956	0.013091	0.02847	
1118.021	77.2776	0.01294	0.012614		1118.021	77.48251	0.012906	0.013598		1120.018	76.34219	0.013099	0.030475		1120.018	50.9596	0.019623	0.025045	
1120.015	77.39035	0.012922	0.012397		1122.021	51.84141	0.01929	0.010622		1122.017	50.60562	0.019761	0.029255		1122.019	44.64786	0.022397	0.024114	
1122.017	51.82941	0.019294	0.009699		1124.022	45.67422	0.021894	0.009824		1124.018	44.83834	0.022302	0.029263		1124.018	44.83408	0.022304	0.023924	
1124.017	45.74318	0.021861	0.009818		1126.021	45.81625	0.021826	0.00992		1126.018	44.85186	0.022296	0.030138		1126.021	44.75419	0.022344	0.023606	
1126.031	45.62486	0.021918	0.009254		1128.022	45.84113	0.021814	0.009636		1128.018	44.9653	0.022239	0.02942		1128.019	44.69627	0.022373	0.024088	
1128.014	45.70019	0.021882	0.008792		1130.023	45.90195	0.021786	0.009579		1130.018	44.83963	0.022302	0.029462		1130.019	44.99155	0.022226	0.024191	
1130.015	45.72579	0.021869	0.010253		1132.021	45.65175	0.021905	0.009521		1132.019	44.8386	0.022302	0.029726		1134.018	53.16101	0.018811	0.02528	
1132.015	45.68931	0.021887	0.009449		1134.021	45.7311	0.021867	0.010103		1134.02	44.90742	0.022268	0.030477		1136.018	71.12058	0.014061	0.027585	
1134.017	45.62658	0.021917	0.00939		1136.022	54.43272	0.018371	0.011171		1136.019	53.83458	0.018575	0.029528		1138.036	71.02432	0.01408	0.027468	
1136.031	53.89666	0.018554	0.011115		1138.022	72.12792	0.013864	0.013233		1138.018	71.39828	0.014006	0.030249		1140.018	71.17631	0.01405	0.027419	
1138.015	72.12714	0.013864	0.011866		1140.021	72.10694	0.013868	0.013128		1140.018	71.38333	0.014009	0.030374		1142.02	71.09584	0.014066	0.02744	
1140.017	72.26564	0.013838	0.012197		1142.023	72.14387	0.013861	0.013089		1142.019	71.36852	0.014012	0.030229		1144.02	81.87289	0.012214	0.028901	
1142.015	71.98267	0.013892	0.012135		1144.023	72.24056	0.013843	0.013457		1144.018	71.1936	0.014046	0.030224		1146.02	103.8052	0.009633	0.031968	
1144.016	72.15477	0.013859	0.012031		1146.021	86.77192	0.011524	0.015563		1146.019	82.55412	0.012113	0.03148		1148.019	104.9745	0.009526	0.032296	
1146.015	86.44498	0.011568	0.014059		1148.023	126.6242	0.007897	0.020599		1148.018	105.4545	0.009483	0.03304		1150.02	103.7775	0.009636	0.032164	
1148.014	127.5284	0.007841	0.017483		1150.023	128.2983	0.007794	0.020348		1150.018	106.1165	0.009424	0.032925		1152.018	104.6991	0.009551	0.032431	
1150.025	125.9313	0.007941	0.017337		1152.044	128.1379	0.007804	0.020713		1152.02	106.6322	0.009378	0.033213		1154.02	92.3651	0.010827	0.030199	
1152.031	125.9421	0.00794	0.017383		1154.023	125.4339	0.007972	0.020061		1154.02	104.8249	0.00954	0.032797		1156.019	69.79826	0.014327	0.02582	
1154.015	126.1817	0.007925	0.017303		1156.02	105.7138	0.009459	0.017796		1156.019	92.07793	0.01086	0.031399		1158.019	69.6651	0.014354	0.026014	
1156.025	102.8034	0.009727	0.015381		1158.021	70.77159	0.01413	0.011339		1158.019	69.71284	0.014345	0.02891		1160.02	69.73068	0.014341	0.026068	
1158.017	70.48469	0.014187	0.011121		1160.022	70.61593	0.014161	0.011491		1160.019	69.86583	0.014313	0.028902		1162.019	69.71954	0.014343	0.02598	
1160.017	70.63969	0.014156	0.01098		1162.019	70.60041	0.014164	0.011952		1162.02	69.9923	0.014287	0.02879		1164.02	61.71277	0.016204	0.025184	
1162.016	70.62227	0.01416	0.010827		1164.023	70.68587	0.014147	0.011603		1164.018	69.82212	0.014322	0.029277		1166.019	44.64197	0.0224	0.022635	
1164.016	70.70076	0.014144	0.010974		1166.024	62.62896	0.015967	0.011307		1166.019	61.70696	0.016206	0.028487		1168.019	44.55585	0.022444	0.022679	
1166.017	62.57905	0.01598	0.010351		1168.043	45.6434	0.021909	0.009915		1168.02	44.7028	0.02237	0.029179		1170.018	44.6232	0.02241	0.022763	
1168.017	45.48673	0.021984	0.008623		1170.022	45.67973	0.021892	0.009987		1170.02	44.73815	0.022352	0.028868		1172.018	44.78547	0.022329	0.022869	
1170.015	45.51474	0.021971	0.009185		1172.032	45.53925	0.021959	0.008902		1172.038	44.65958	0.022392	0.029856		1174.018	44.60062	0.022421	0.023083	
1172.016	45.40508	0.022024	0.009851		1174.022	45.74146	0.021862	0.009846		1174.044	44.60321	0.02242	0.029139		1176.019	44.83242	0.022305	0.02266	
1174.016	45.64225	0.02191	0.009553		1176.021	45.8151	0.021827	0.009308		1176.02	44.7688	0.022337	0.028827		1178.02	44.66832	0.022387	0.023556	
1176.034	45.51683	0.02197	0.008528		1178.022	45.5057	0.021975	0.009092		1178.018	44.79334	0.022325	0.029496		1180.02	70.42811	0.014199	0.02678	
1178.015</																			

VLBT#	56			VLBT#	57		
SURFACE	HVPC ctg			SURFACE	HVPC ctg		
		Ave COF	0.006			Ave COF	0.007
		StdDev	0.001			StdDev	0.001
		Max COF	0.009			Max COF	0.010
		Min COF	0.005			Min COF	0.005
Time	Load, P	1/P	Fric Coeff	Time	Load, P	1/P	Fric Coeff
(s)	(N)	(N ⁻¹)	(n/d)	(s)	(N)	(N ⁻¹)	(n/d)
1080.018	36.59672	0.027325	0.007628	1080.038	52.7898	0.018943	0.00599
1082.02	32.52089	0.030749	0.008367	1082.023	33.35009	0.029985	0.008509
1084.019	32.54778	0.030724	0.009075	1084.023	29.16408	0.034289	0.009736
1086.018	32.44983	0.030817	0.009461	1086.023	29.27072	0.034164	0.008915
1088.019	32.61433	0.030661	0.007948	1088.023	29.20188	0.034244	0.010386
1090.012	32.72129	0.030561	0.008387	1090.027	28.85204	0.03466	0.009779
1092.018	32.48682	0.030782	0.008974	1092.021	29.11294	0.034349	0.009906
1094.02	32.3891	0.030875	0.009435	1094.022	29.12649	0.034333	0.009488
1096.02	32.62658	0.03065	0.00814	1096.025	29.19542	0.034252	0.009823
1098.018	32.59324	0.030681	0.008936	1098.024	29.05468	0.034418	0.00981
1100.019	52.49177	0.019051	0.005379	1100.024	29.24297	0.034196	0.010028
1102.019	57.19178	0.017485	0.006475	1102.024	49.65892	0.020137	0.006568
1104.02	57.39909	0.017422	0.005817	1104.024	53.8656	0.018565	0.005329
1106.019	57.39645	0.017423	0.005777	1106.025	53.75651	0.018602	0.006568
1108.02	57.23058	0.017473	0.005842	1108.023	53.87547	0.018561	0.006195
1110.019	74.56093	0.013412	0.005211	1110.024	54.01417	0.018514	0.005911
1112.021	77.93781	0.012831	0.004968	1112.026	70.96283	0.014092	0.005751
1114.02	78.04081	0.012814	0.004831	1114.025	74.24463	0.013469	0.00568
1116.019	77.9418	0.01283	0.004833	1116.023	74.2062	0.013476	0.005657
1118.02	77.83656	0.012847	0.004798	1118.05	74.2248	0.013473	0.005937
1120.02	52.33795	0.019107	0.006297	1120.024	74.03972	0.013506	0.005662
1122.018	46.23881	0.021627	0.005923	1122.025	48.86575	0.020464	0.006694
1124.02	46.24986	0.021622	0.006952	1124.024	42.69431	0.023422	0.007843
1126.02	46.41315	0.021546	0.007274	1126.041	42.77433	0.023379	0.006367
1128.02	46.47353	0.021518	0.0066	1128.024	42.75452	0.023389	0.007491
1130.021	46.25534	0.021619	0.006596	1130.035	42.80486	0.023362	0.00692
1132.02	46.15851	0.021664	0.007128	1132.025	42.70912	0.023414	0.006749
1134.02	54.89241	0.018217	0.005689	1134.026	42.78291	0.023374	0.00733
1136.018	72.88418	0.01372	0.004934	1136.024	51.53756	0.019403	0.005843
1138.019	72.86142	0.013725	0.004919	1138.024	70.33978	0.014217	0.005616
1140.018	72.6834	0.013758	0.005037	1140.024	70.22564	0.01424	0.00626
1142.028	72.73747	0.013748	0.004927	1142.023	70.23124	0.014239	0.005734
1144.019	92.70528	0.010787	0.006127	1144.047	70.34443	0.014216	0.005575
1146.02	151.9346	0.006582	0.009036	1146.024	79.04606	0.012651	0.006454
1148.02	147.5769	0.006776	0.008632	1148.025	99.75572	0.010024	0.008124
1150.02	145.8858	0.006855	0.008338	1150.025	99.61036	0.010039	0.008068
1152.02	144.5056	0.00692	0.008318	1152.024	99.71847	0.010028	0.008261
1154.021	115.0827	0.008689	0.007611	1154.024	98.48012	0.010154	0.00813
1156.019	71.32516	0.01402	0.00491	1156.033	89.00965	0.011235	0.008099
1158.018	71.3862	0.014008	0.005135	1158.024	67.51138	0.014812	0.006116
1160.02	71.27737	0.01403	0.004804	1160.024	67.52659	0.014809	0.006538
1162.019	71.22741	0.01404	0.004792	1162.023	67.55021	0.014804	0.005679
1164.019	63.16135	0.015832	0.005566	1164.025	67.63496	0.014785	0.006074
1166.019	46.07235	0.021705	0.00649	1166.025	59.68869	0.016754	0.006194
1168.02	46.38382	0.021559	0.006488	1168.051	42.43852	0.023563	0.006896
1170.02	46.1538	0.021667	0.006771	1170.026	42.60934	0.023469	0.007145
1172.021	46.01582	0.021732	0.006643	1172.024	42.60449	0.023472	0.007455
1174.019	46.20325	0.021644	0.006204	1174.026	42.44182	0.023562	0.007492
1176.019	46.19986	0.021645	0.006979	1176.024	42.57499	0.023488	0.007245
1178.019	46.18061	0.021654	0.006951	1178.024	42.59539	0.023477	0.00717
1180.019	71.64799	0.013957	0.00458	1180.023	42.61189	0.023468	0.007225
1182.02	78.12119	0.012801	0.005185	1182.025	68.22145	0.014658	0.006463
1184.019	77.86873	0.012842	0.004885	1184.027	73.87462	0.013536	0.006115
1186.02	77.71049	0.012868	0.004667	1186.026	74.07531	0.0135	0.006394
1188.021	77.74742	0.012862	0.00516	1188.025	73.94401	0.013524	0.006277
1190.019	59.17474	0.016899	0.005837	1190.026	73.73637	0.013562	0.006157
1192.021	55.79567	0.017923	0.005479	1192.047	56.01297	0.017853	0.006804
1194.019	55.72421	0.017946	0.005756	1194.023	52.63402	0.018999	0.006584
1196.02	55.82434	0.017913	0.005618	1196.039	52.49492	0.019049	0.006071
1198.019	55.92169	0.017882	0.006273	1198.024	52.5735	0.019021	0.006609
1200.02	36.80502	0.02717	0.007357	1200.024	52.67965	0.018983	0.006463
1202.044	32.40931	0.030855	0.007696	1202.025	33.37196	0.029965	0.008879
1204.021	32.60882	0.030667	0.008388	1204.025	29.06504	0.034406	0.010192
1206.02	32.5196	0.030751	0.008463	1206.027	29.04917	0.034424	0.010674

Appendix C. Selected Properties of Cast Iron, Bronze, and Titanium Alloy

Property	Units	Gray Cast Iron*	Alloy 932 Bronze**	Ti64 (Grade 5)***
Nominal composition	wt%	3.4 – 3.7 wt% C, resid. S, P, O, Si, bal Fe.	83 Cu, 7.0 Pb, 6.9 Sn, 2.5 Zn, 0.35 Sb, 0.2 Fe, 1.0 Ni, minor Al, Si, S	5.5 – 6.8 Al, 3.5 – 4.5 V, 0.3 Fe, 0.2 O, 0.1 C, 0.05 N, ~ 0.01 H
Density	g/cm ³	7.2	8.91	4.42
Melting point	°C		854	1650 - 1670
Elastic modulus	GPa	~ 200. – 211.	100.	107 – 122 (20 C) 105 – 111 (230 C)
Poissons ratio	(none)	0.17		0.31
Thermal conductivity	W/m-°K	42 – 62	58.2 (20 C)	6.6 (20 C) 10.6 (315 C)
Specific heat	J/(kg - °K)	460.	377.	586. (20 – 570 C)
Coefficient of thermal expansion	x 10 ⁻⁶ / °K	8.1 – 19.3 (20 C)	18.0 (20-100 C)	9.0 (0 - 100 C) 9.4 (20 – 425 C)

Table references:

* R. L. Hecht, R. B. Dinwiddie, W. D. Porter, and H. Wang (1996) “Thermal Transport Properties of Grey Cast Irons,” Soc. of Automotive Engineers, paper number 962126.

** Atlas Bronze website, www.atlasbronze.com, on-line data table (2012).

*** TIMET Corporation, Denver, CO, on-line alloy data sheets (2002).

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