

A Viscoplastic Constitutive Model for Active Braze Alloys

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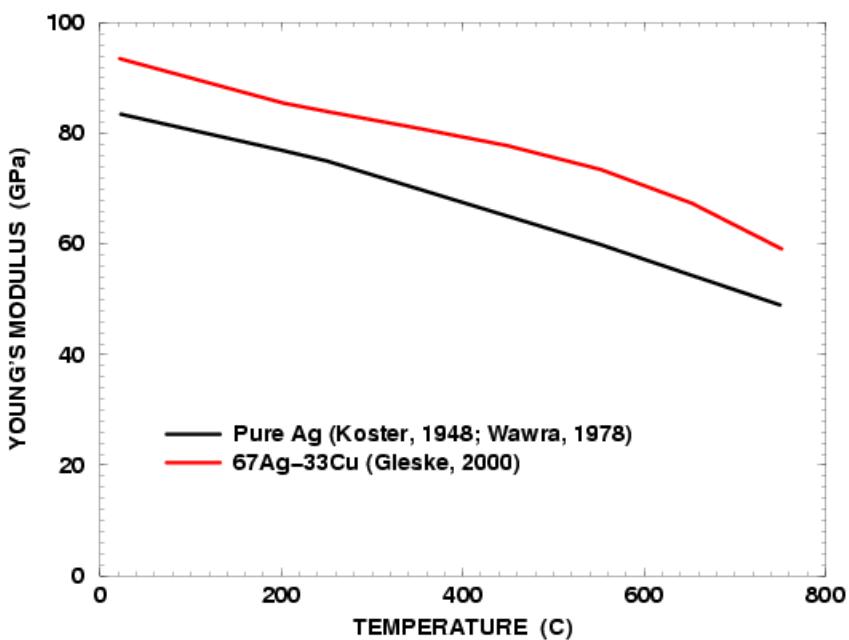


Outline

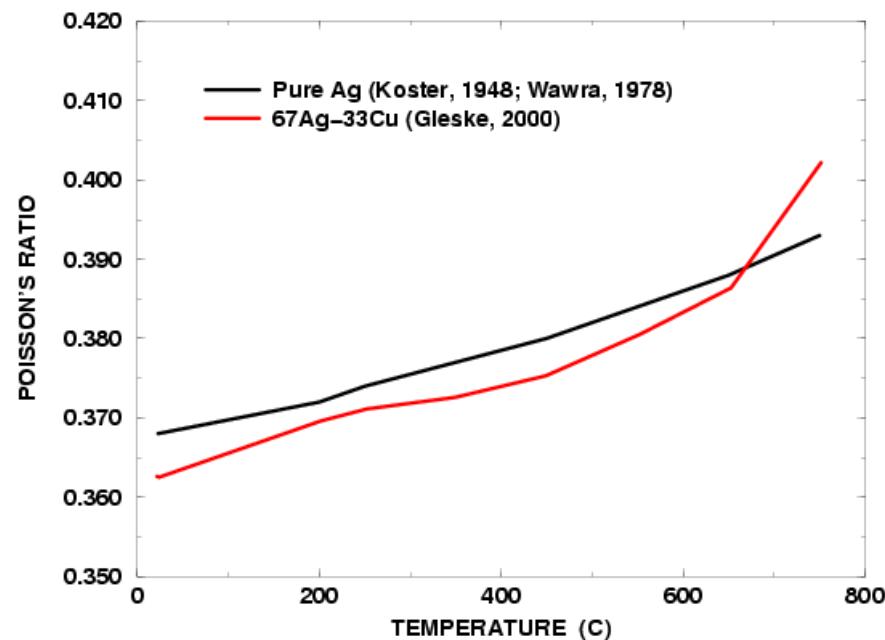
- Mechanical Behavior of Three Active Braze Alloys
63Ag-35.25Cu-1.75Ti, 98Ag-2Zr (wt%), 97Ag-1Cu-2Zr
- Behavior Captured with Existing Models
- Development of Viscoplastic Constitutive Model
- Simulation of Metal-to-Ceramic Brazing

Elastic Moduli for Pure Ag and 67Ag-33Cu

Young's Modulus



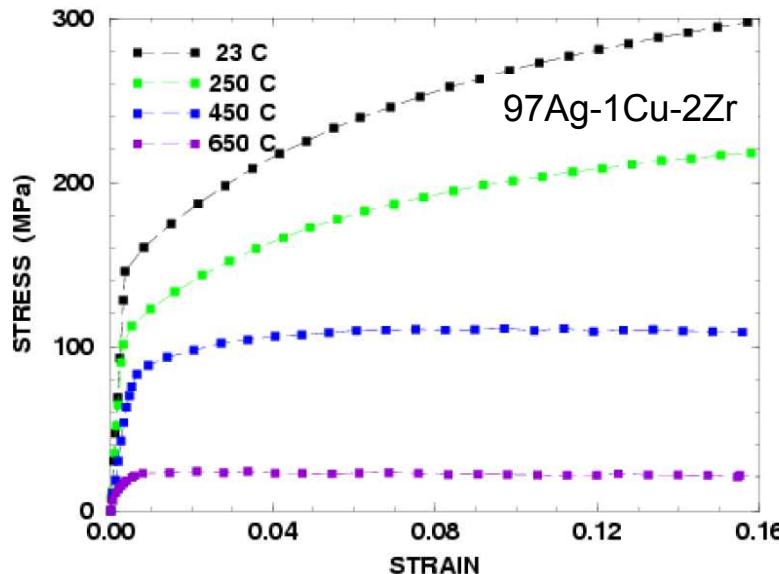
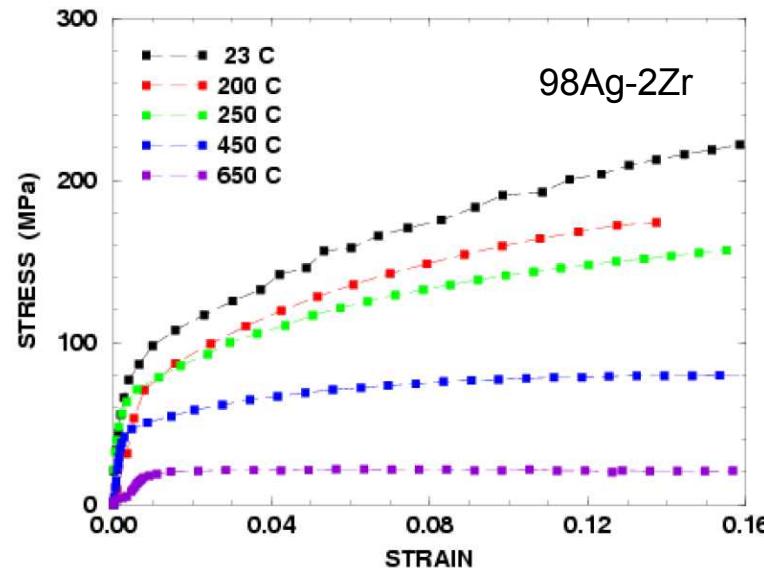
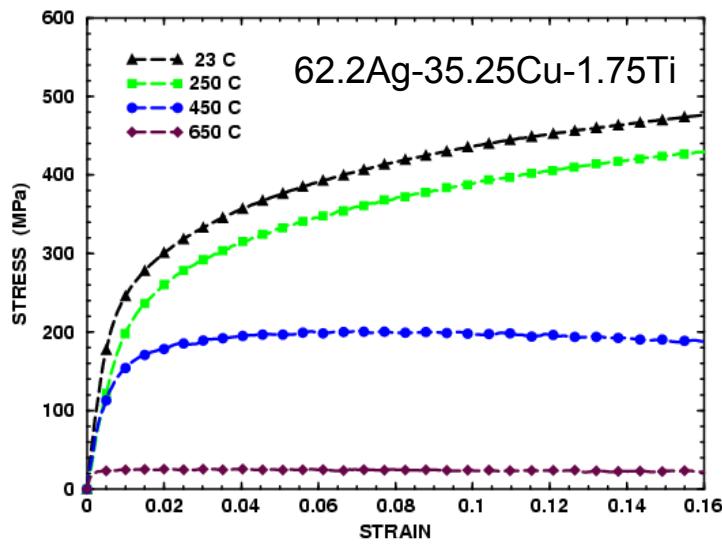
Poisson's Ratio



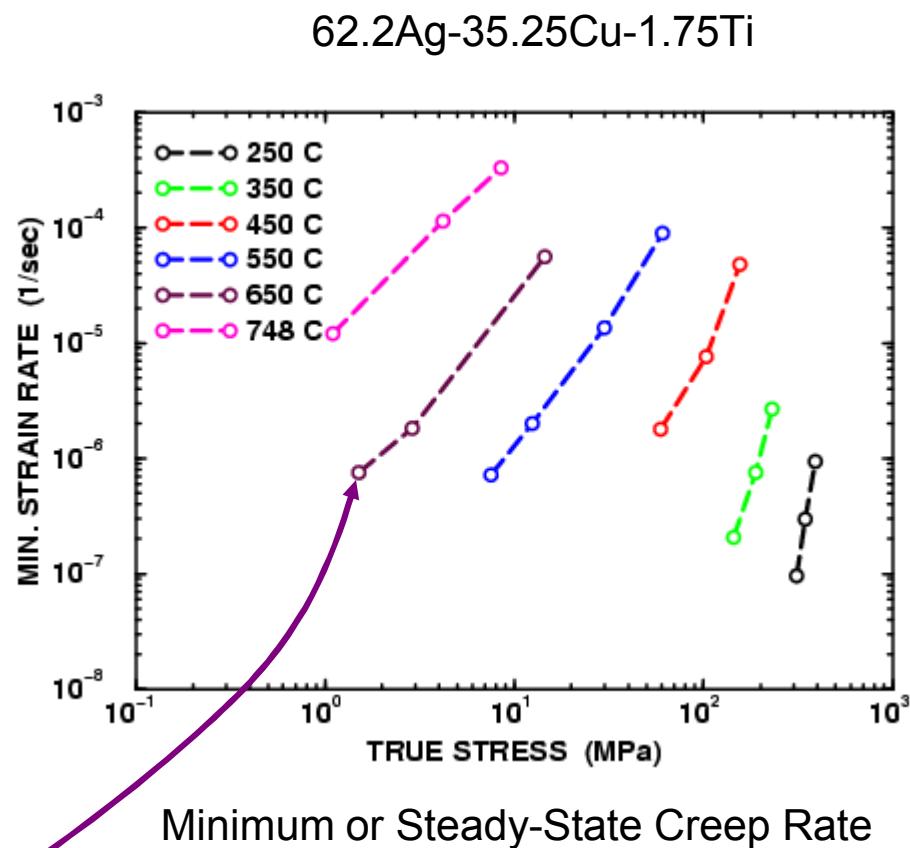
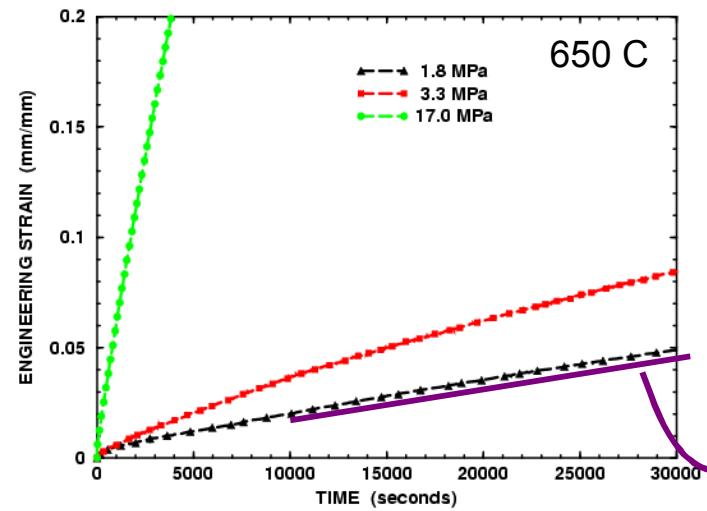
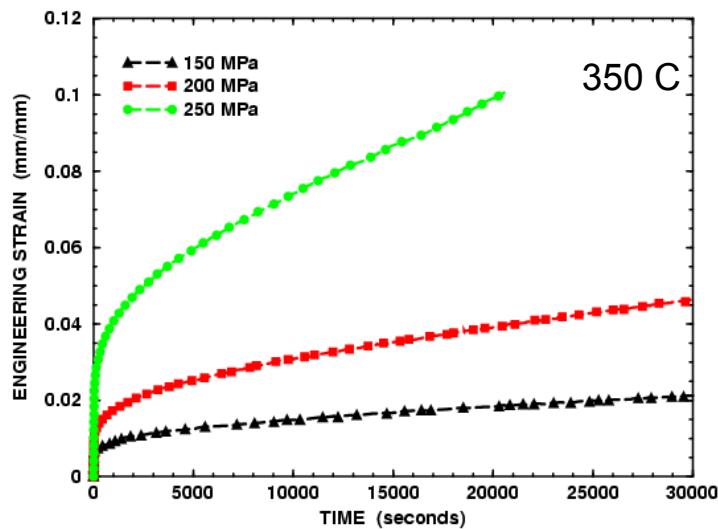
References:

- Koster, W., *Z. Metallk.*, **39**, pp. 145, 1948.
- Wawra, H., *Z. Metallk.*, **69**, pp. 518-523, 1978
- Gieske, J., internal memo, Sandia National Labs, 2000.

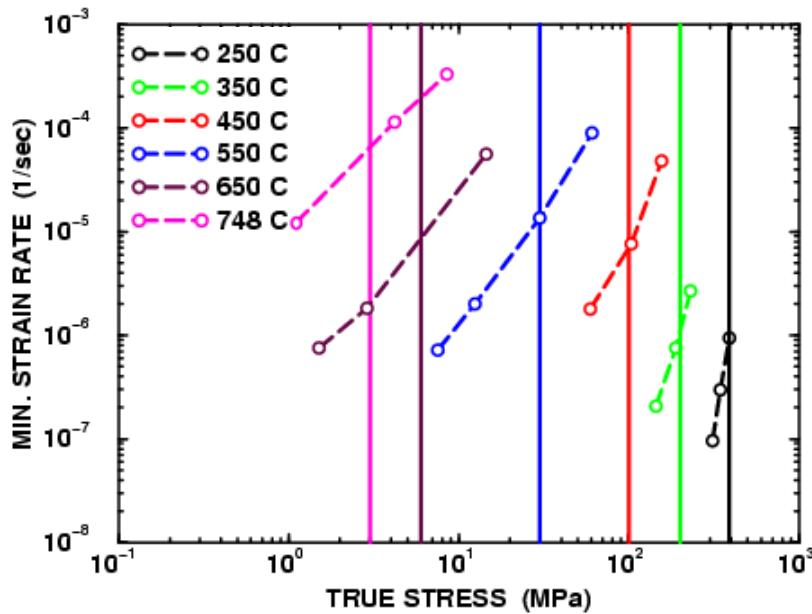
Isothermal Uniaxial Compression Experiments (constant true strain rate of 1.67×10^{-4} per sec.)



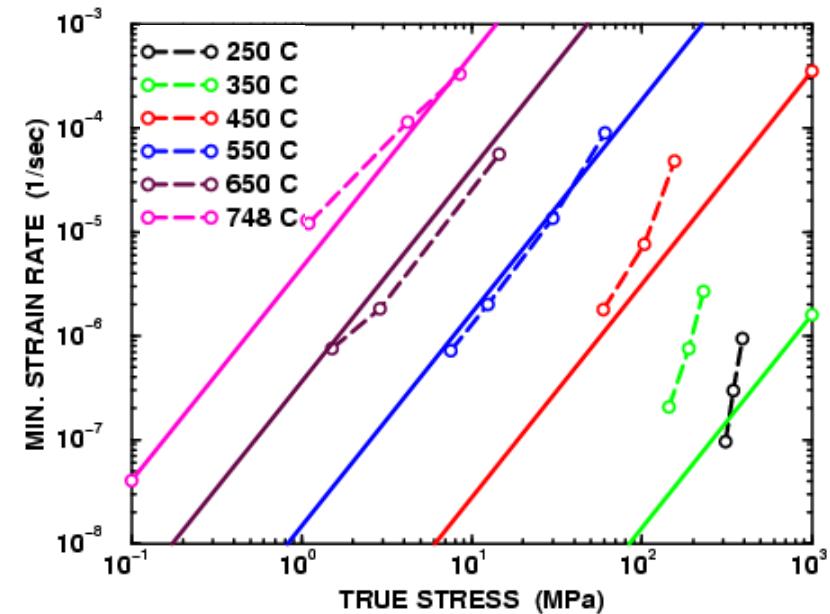
Creep Compression Experiments (constant engineering stress)



Steady-State Creep vs. Stress



vonMises Plasticity Model

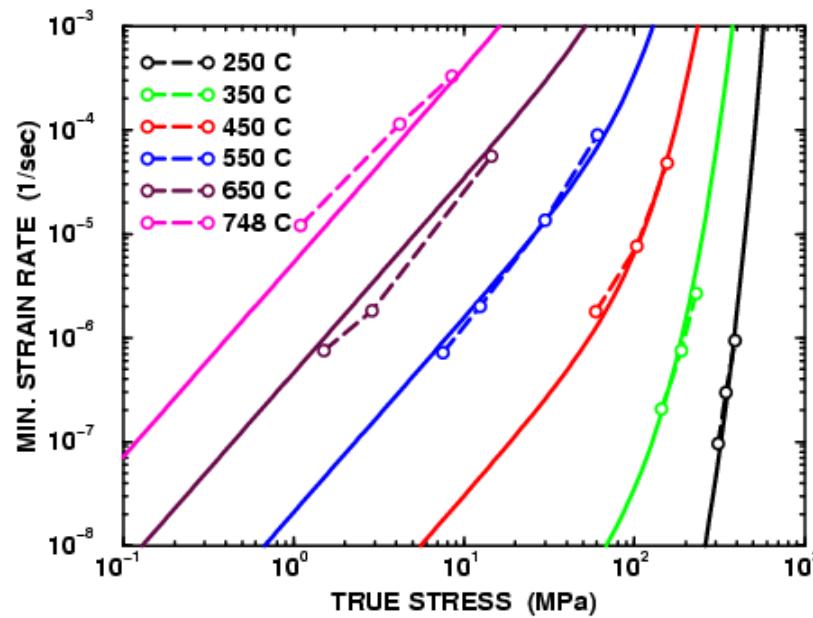


Power-Law Secondary Creep Model

$$\dot{\epsilon}^{ss} = 9.9388 \times 10^4 (\sigma)^{2.05} \exp\left(\frac{-48,300}{R\theta}\right)$$

R is the gas constant (1.987 cal/mole/Kelvin)

Steady-State Creep vs. Stress



Hyperbolic Sine Secondary Creep Model

$$\dot{\epsilon}^{ss} = 8.13 \times 10^7 \sinh^{1.87} \left(\frac{\sigma}{50.66} \right) \exp \left(\frac{-46,706}{R\theta} \right)$$

Ref: Garofalo, F. **Fundamentals of Creep and Creep Rupture in Metals**, MacMillan Company (1965)



A Viscoplastic (UCP) Model

Total strain rate = elastic + inelastic

$$\dot{\boldsymbol{\varepsilon}} = \dot{\boldsymbol{\varepsilon}}^e + \dot{\boldsymbol{\varepsilon}}^{in}$$

Isotropic, linear elastic

$$\boldsymbol{\sigma} = \mathbf{E} : \boldsymbol{\varepsilon}^e$$

Inelastic strain rate

$$\dot{\boldsymbol{\varepsilon}}^{in} = \frac{3}{2} \gamma \mathbf{n} = \frac{3}{2} f \sinh^p \left(\frac{\tau}{D} \right) \mathbf{n}$$

Normalized stress difference tensor
 \mathbf{s} is the stress deviator, \mathbf{B} is state tensor

$$\mathbf{n} = \frac{\mathbf{s} - \frac{2}{3} \mathbf{B}}{\tau}$$

Effective Stress: scalar measure of stress difference magnitude

$$\tau = \sqrt{\frac{3}{2} \left(\mathbf{s} - \frac{2}{3} \mathbf{B} \right) : \left(\mathbf{s} - \frac{2}{3} \mathbf{B} \right)}$$

References:

Miller, A.K., **Unified Constitutive Equations for Creep and Plasticity**, Elsevier, 1987.

Neilsen, M.K. et al., SAND96-0984, Sandia National Labs, 1996.

Johnson, G.C. and Bammann, D.J., *Intl. J. Solids Structures*, **20**, 725-737, 1984.

Flanagan, D.P., and Taylor, L.M., *Comp. Mth. Appl. Mech. Engr.*, **62**, 305-320, 1987.



A Viscoplastic (UCP) Model (cont.)

Evolution Eq. for State Variable D

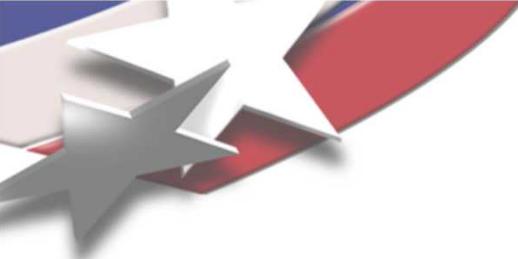
$$\dot{D} = \frac{A_1 \gamma}{(D - D_0)^{A_3}} - A_2 (D - D_0)^2$$

Evolution Eq. for State Tensor \mathbf{B}

$$\dot{\mathbf{B}} = \frac{A_4 \mathbf{d}^{in}}{b^{A_6}} - A_5 b \mathbf{B}$$

Magnitude of \mathbf{B}

$$b = \sqrt{\frac{2}{3} \mathbf{B} : \mathbf{B}}$$

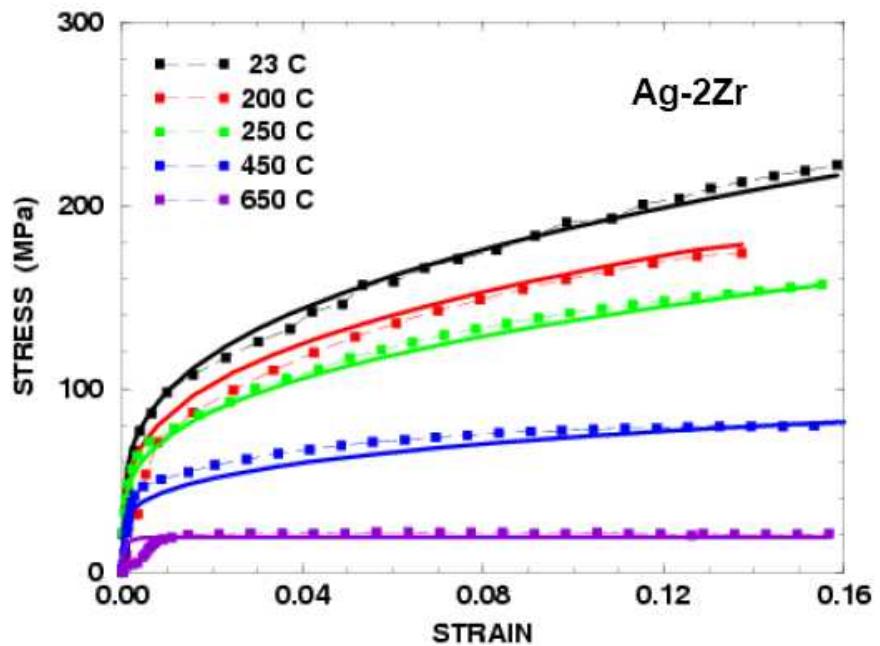
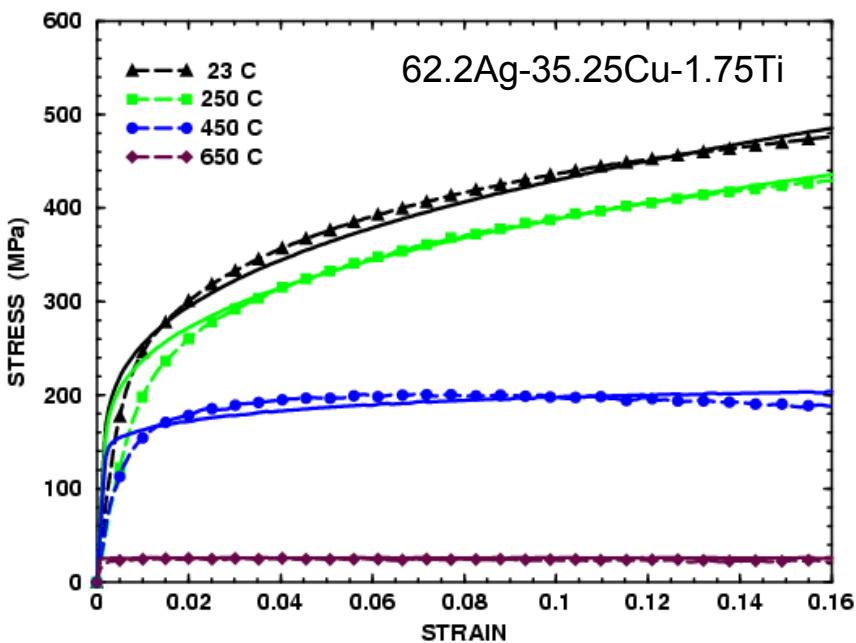


Material Parameters for UCP Model

Table 3. 98Ag-2Zr Material Parameters.

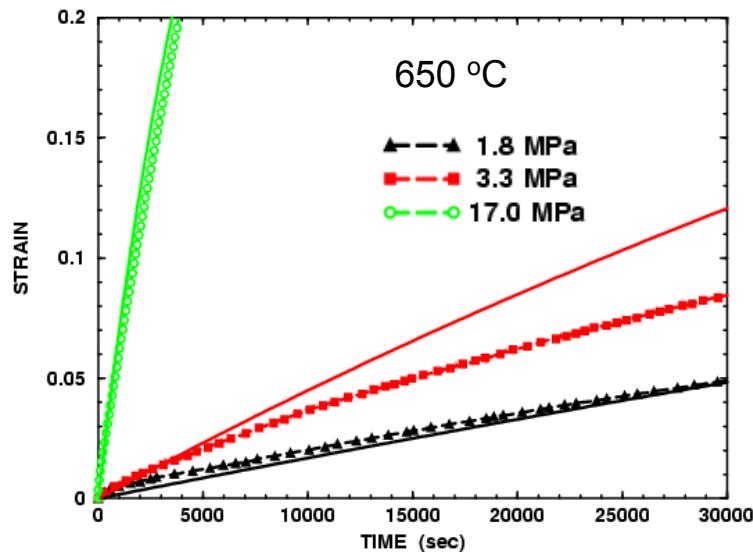
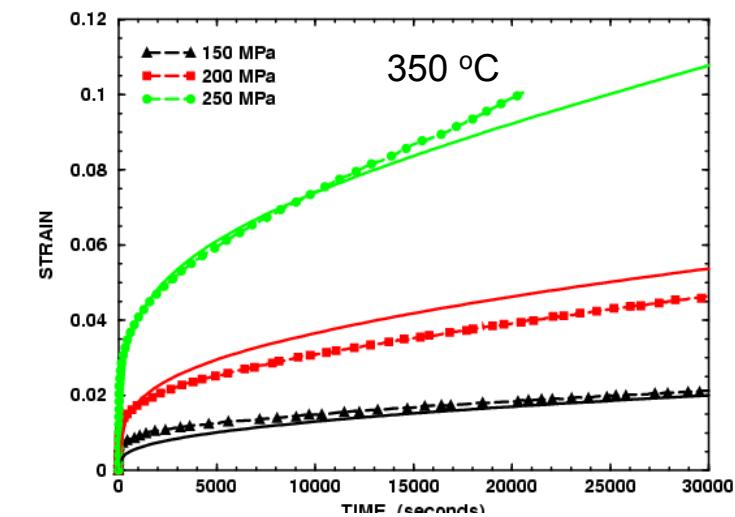
Temp. (°C)	Flow Rate $\ln(f)$	Sinh Expo. p	Isotropic Hardening, A_1 (MPa^{A_3+1})	Isotropic Recov., A_2 $1/(\text{MPa}\cdot\text{sec})$
23	-86.29	12.0	15840.	2.117e-11
200	-71.11	11.4	12300.	2.000e-10
250	-64.10	11.26	12290.	8.469e-8
350	-37.15	5.81	3933.	3.191e-7
450	-30.36	5.81	3245.	9.641e-6
550	-24.59	5.81	3100.	2.541e-4
650	-18.12	5.81	3087.	6.457e-3
750	-10.67	5.81	3000.	7.116e-3
<hr/>				
Iso. Exponent, A_3		1.7278		
Kin. Harden., A_4 (MPa^{A_6+1})		0.0		
Kin. Rec., A_5 $1/(\text{MPa}\cdot\text{sec})$		0.0		
Kin. Exponent, A_6		1.7278		
Flow Stress, D_0 (MPa)		5.00		

Simulations of Uniaxial Compression Experiments with Viscoplastic Model Show Good Agreement with Data

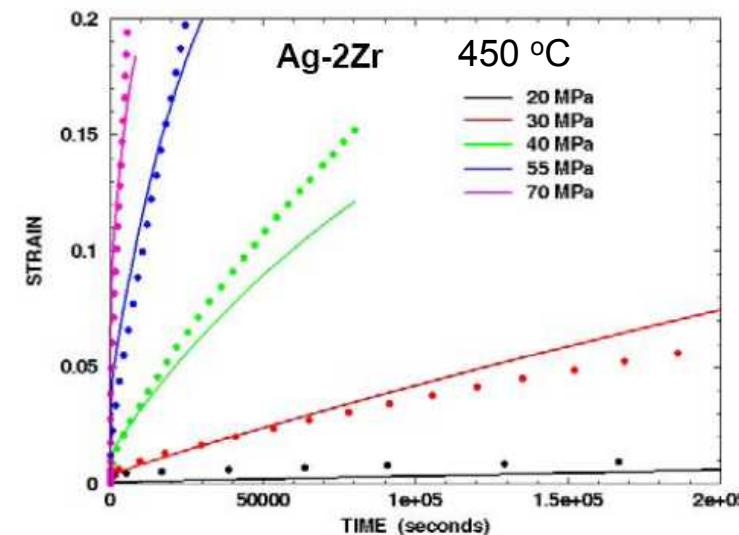
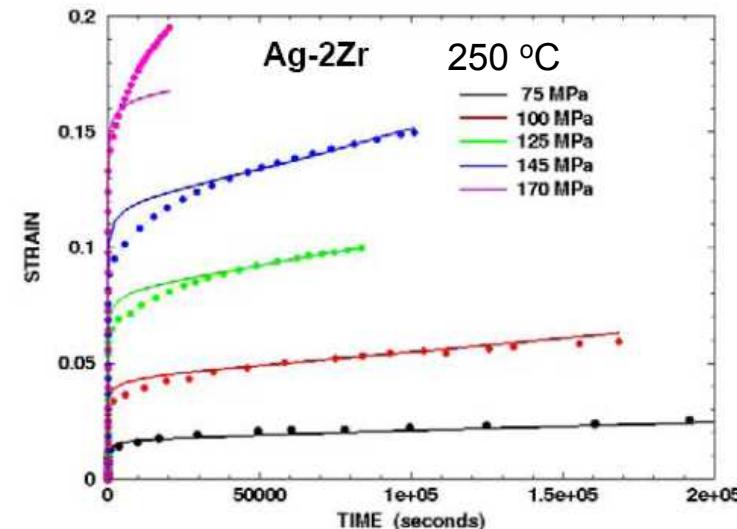


- solid lines show model predictions
- symbols show experimental results

Simulation of Creep Compression Experiments

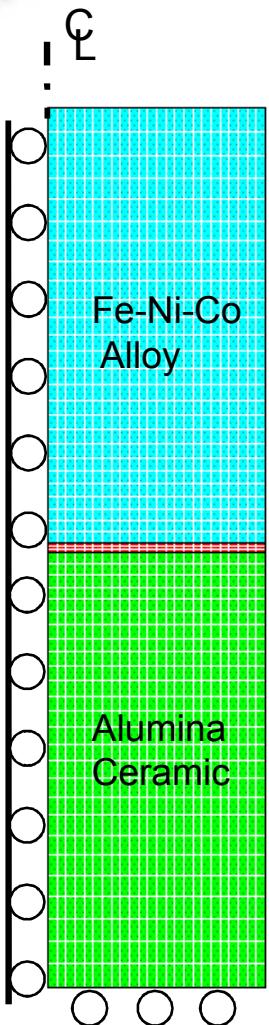


62.2Ag-35.25Cu-1.75Ti

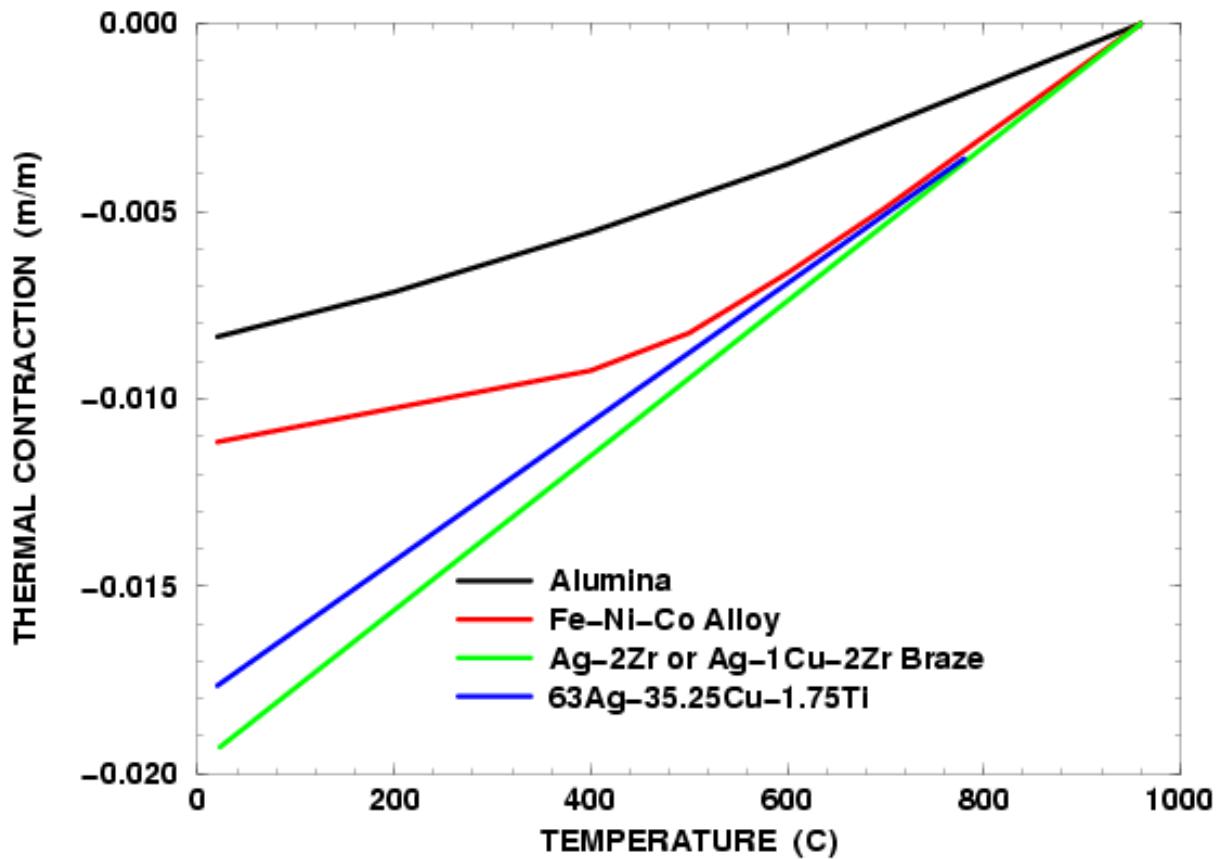


- solid lines show model predictions
- symbols show experimental results

Simulation of Metal-to-Ceramic Brazing

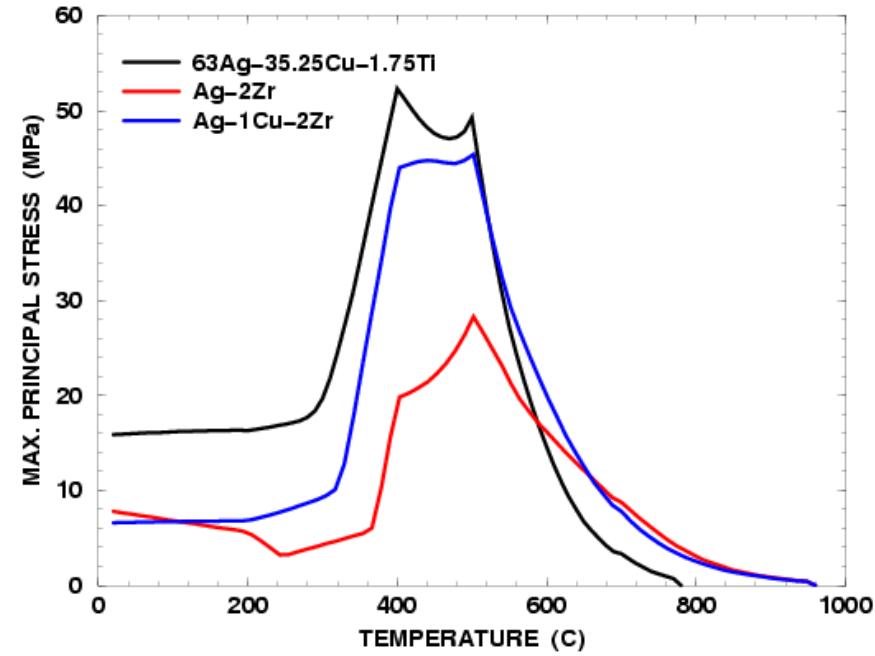
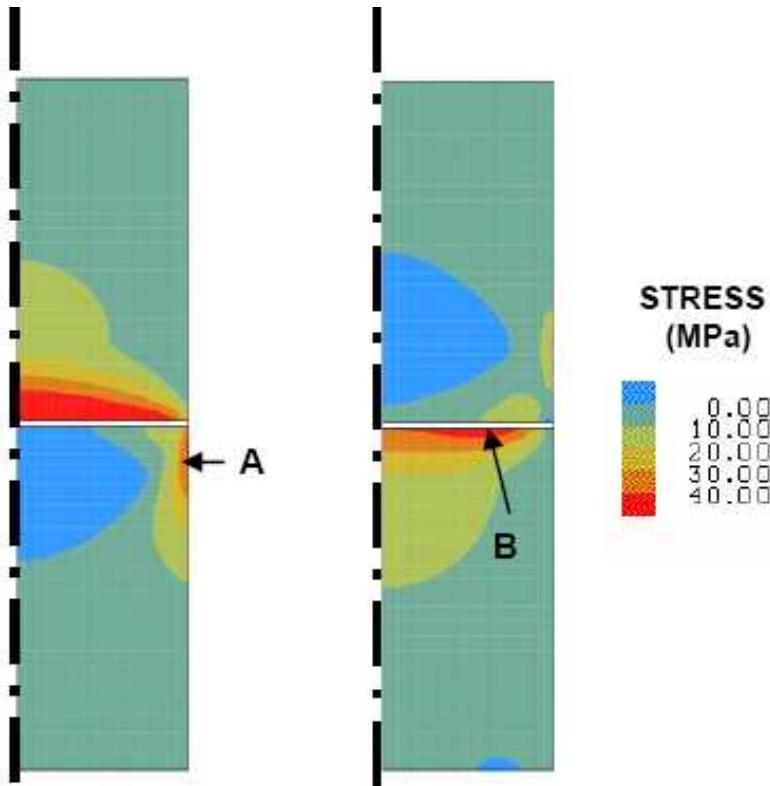


Axisymmetric finite element model

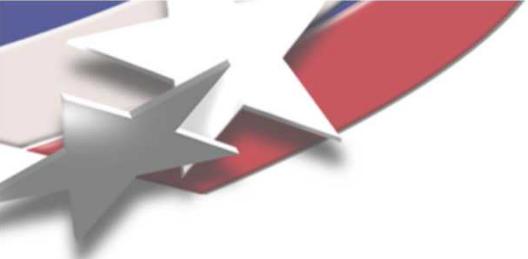


Thermal expansion of braze joint materials

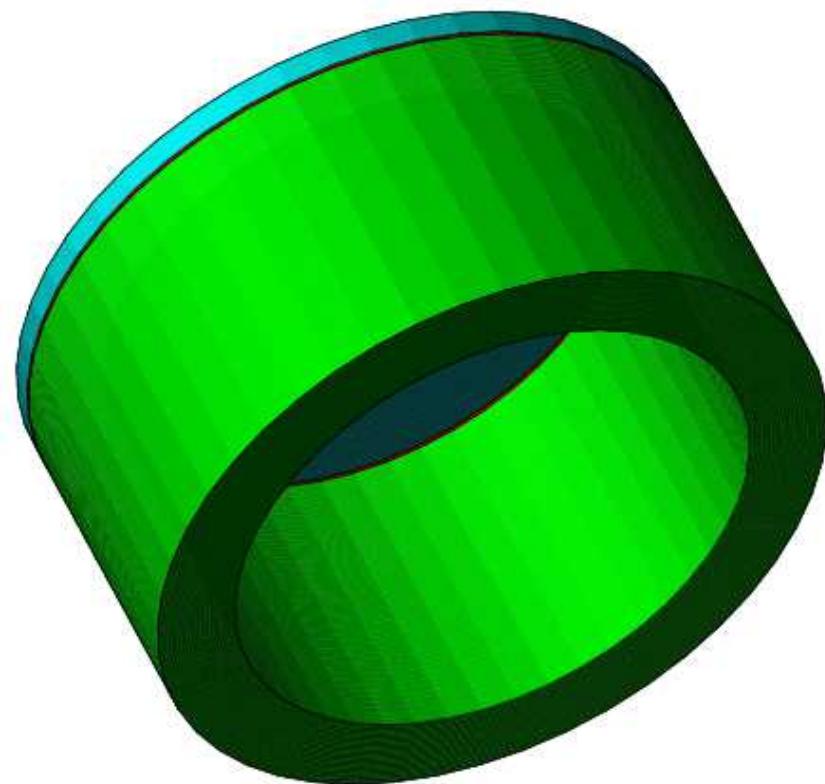
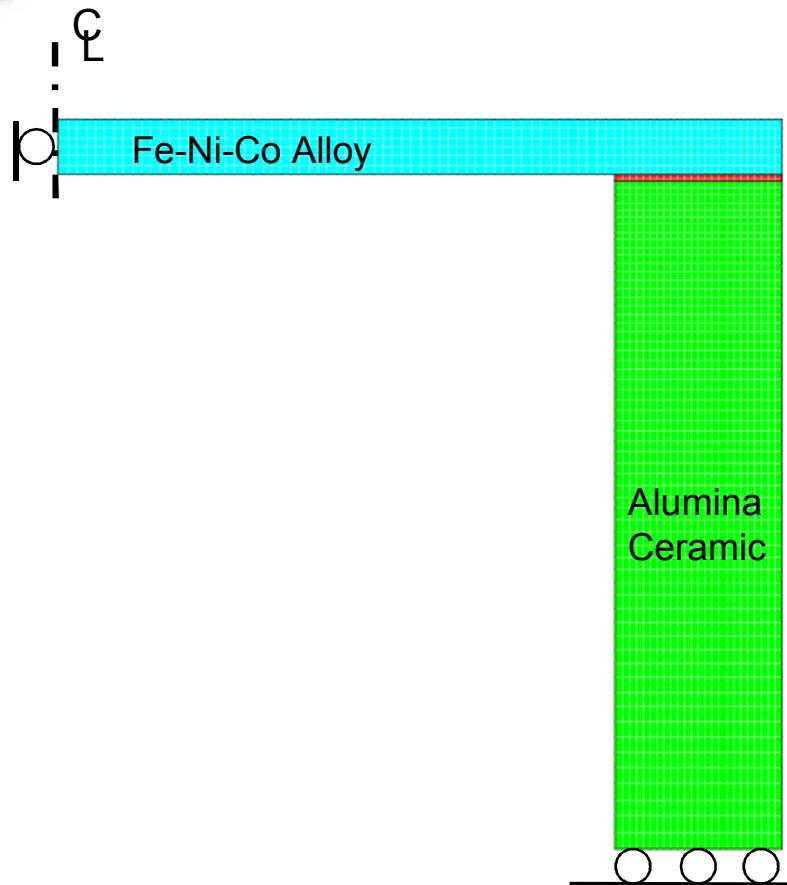
Simulation of Metal-to-Ceramic Brazing



Maximum tensile stress distribution in alumina ceramic during post-braze cooldown cycle, 97Ag-1Cu-2Zr

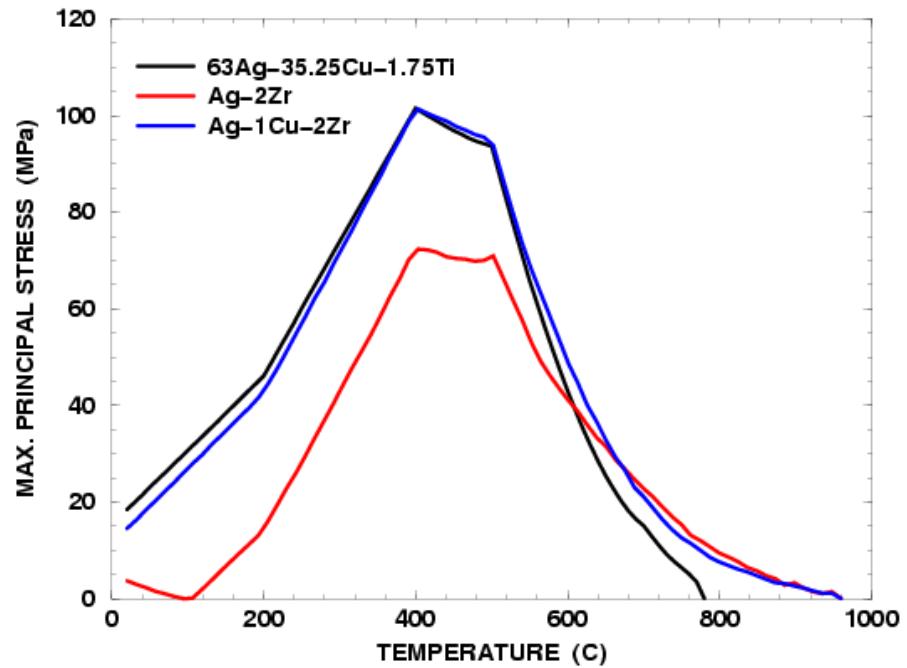
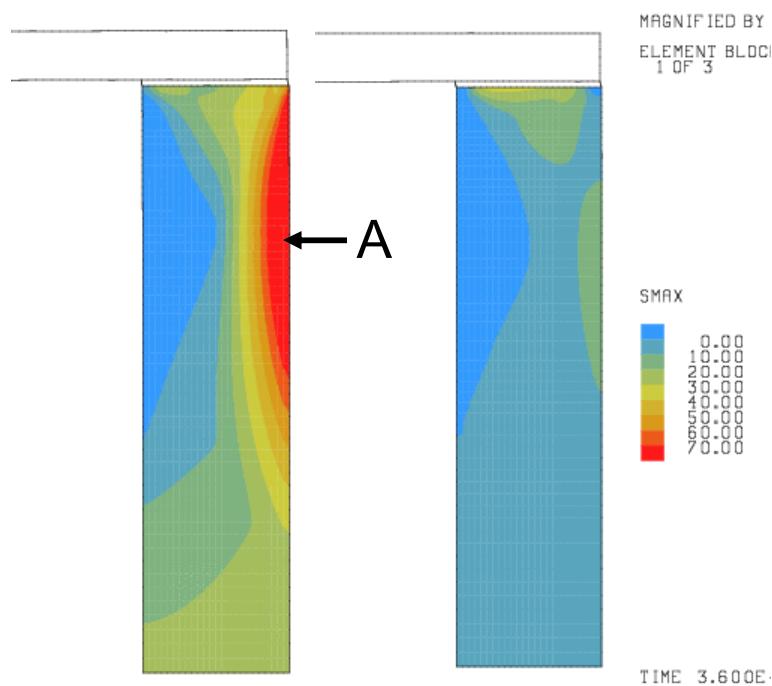


Simulation of Metal-to-Ceramic Brazing



Axisymmetric finite element model
20.0 mm I.D. 26.0 mm O.D, 1.0 mm thick washer

Simulation of Metal-to-Ceramic Brazing

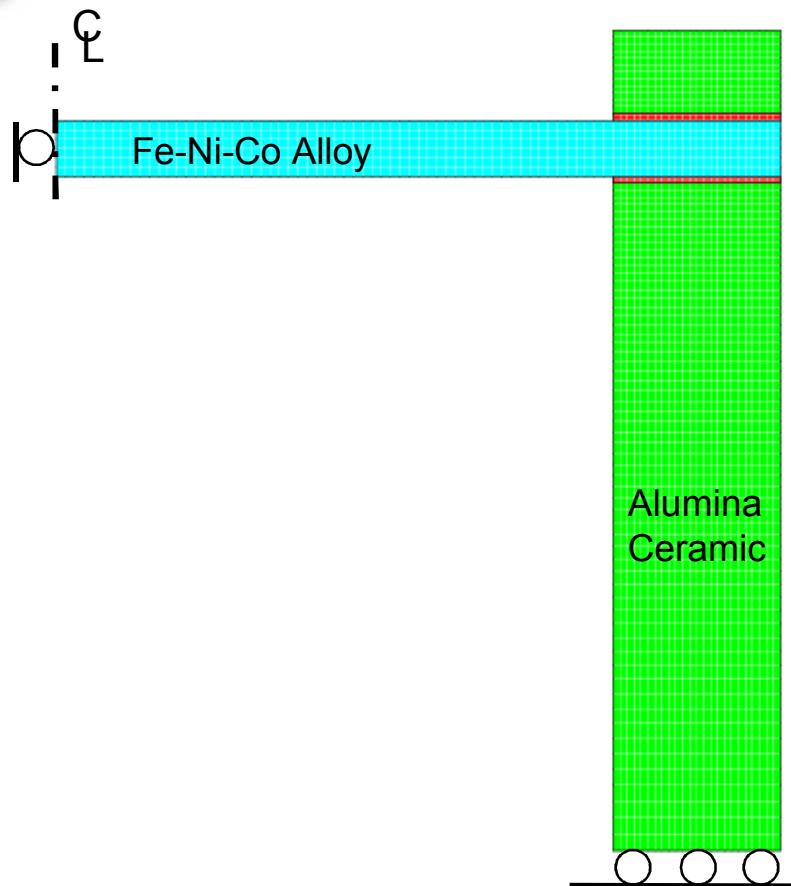


Maximum Tensile Stress History at Point A

Maximum tensile stress generated in alumina ceramic during post-braze cooldown cycle, 97Ag-1Cu-2Zr

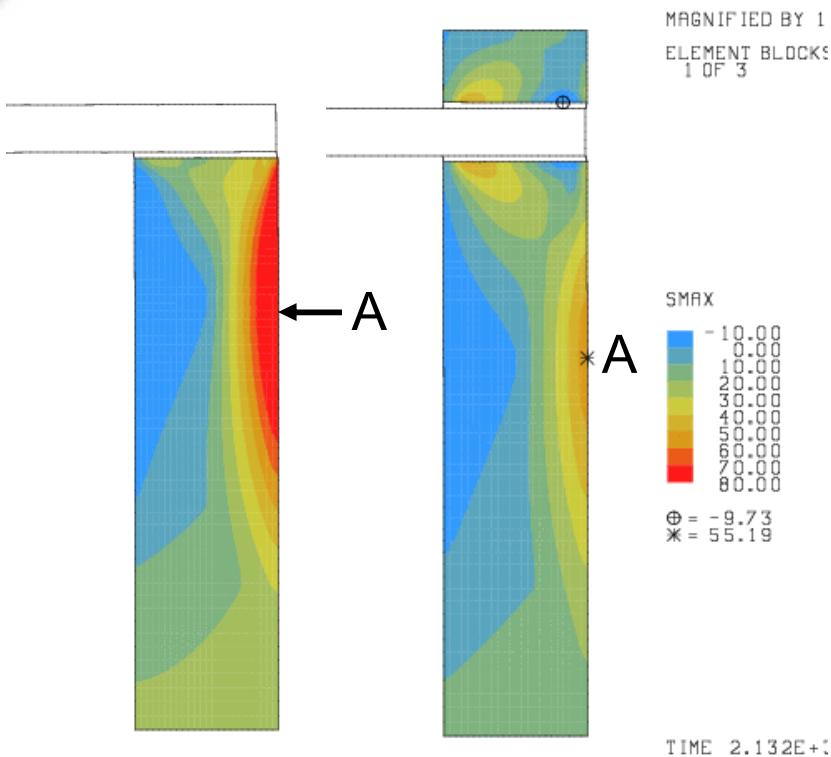


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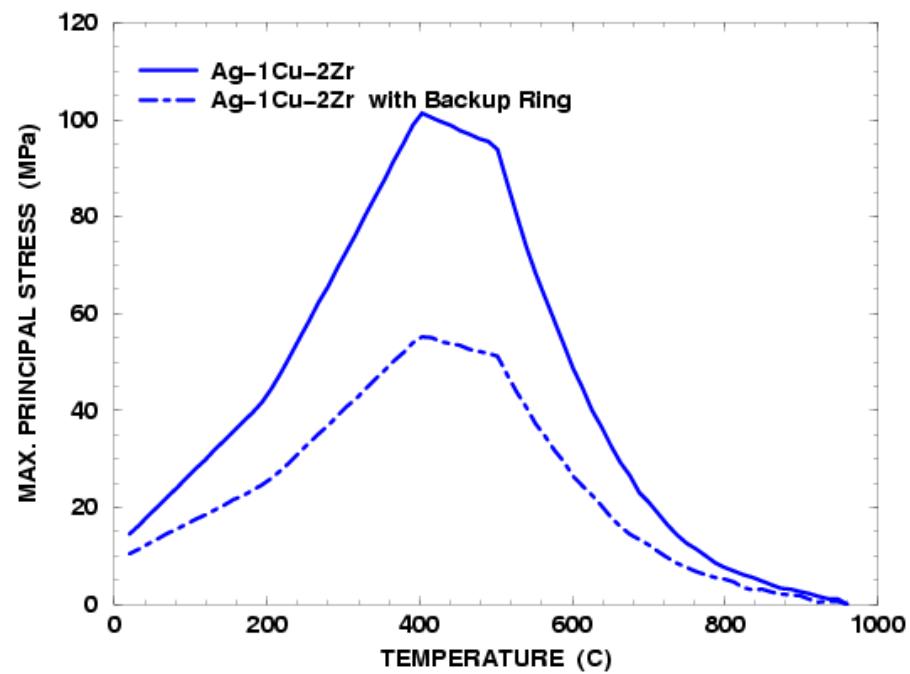


Axisymmetric finite element model
20.0 mm I.D. 26.0 mm O.D, 1.0 mm thick washer

Simulation of Metal-to-Ceramic Brazing

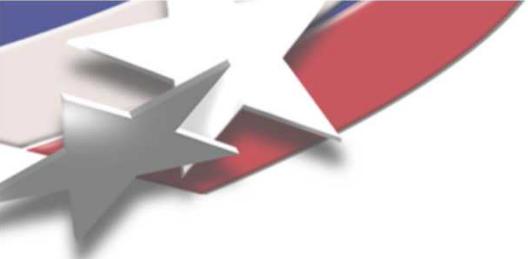


w/o backup ring with backup ring
(403.0 °C) (403.0 °C)



Maximum Tensile Stress History at Point A

Maximum tensile stress generated in alumina ceramic during post-braze cooldown cycle, 97Ag-1Cu-2Zr



Summary

- Viscoplastic models were developed for 3 active braze alloys: 63Ag-35.25Cu-1.75Ti, 98Ag-2Zr, 97Ag-1Cu-2Zr
- Parameters for the models were obtained from a combination of uniaxial compression and creep compression experiments
- Cyclic loading experiments are needed to identify both isotropic and kinematic hardening/recovery parameters.
- The models were used to investigate stress generated during metal-to-ceramic brazing.
- The peak ceramic stress generated during brazing of a Fe-Ni-Co alloy rod to alumina ceramic rod was highest when the lower temperature 63Ag-35.25Cu-1.75Ti braze alloy was used.
- Backup rings were shown to be effective in reducing both transient and residual tensile stress levels in the ceramic.