



# Understanding Surface Reconstruction Coexistence in III-V Alloys

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**Normand Modine**

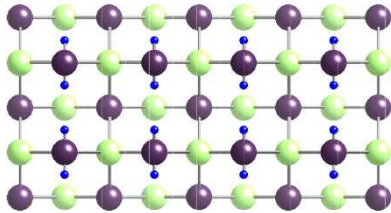
*Sandia National Laboratories - CINT, Albuquerque NM*

# Semiconductor Surfaces Reconstruct to Lower Energy

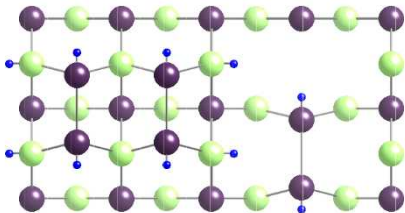
## InAs (001)

### Unreconstructed

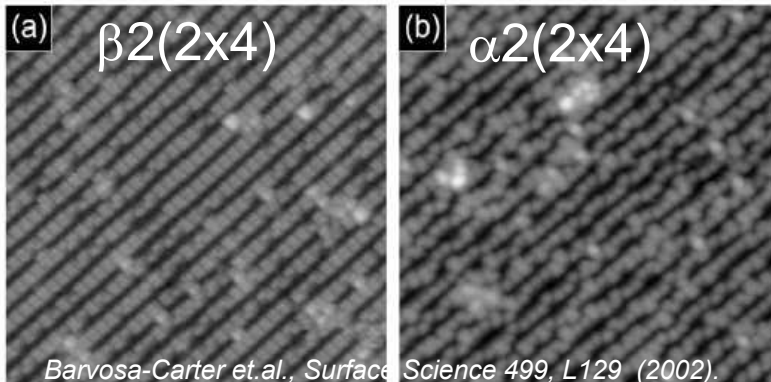
- = Anion
- = Cation
- = Dangling Bond



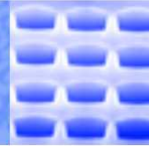
### $\beta 2(2 \times 4)$



### Scanning Tunneling Microscopy (STM) Images



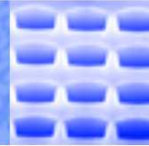
- When a surface is formed, the dangling bonds rearrange on the surface to minimize their energy
  - Competing Interactions
    - Local Chemistry
    - Long Range Electrostatics
    - Strain
- Surface reconstructions affect
  - Ordering in alloys
  - Interfaces
  - Self Assembly
  - Catalysis
  - Epitaxial Film Growth
- InAs (001) and GaAs (001)
  - Single Reconstruction Surface
  - Changes with Chemical Potential



## Outline

Goal: Examine Atomic Surface Structure of III-V Alloys

- Ternary alloys exhibit reconstruction *coexistence*
- Understanding the Role of Strain
  - Atomic size mismatch strain
    - $z(4 \times 4)$  reconstruction
    - $(4 \times 3)$  reconstruction (?)
  - Lattice mismatch strain
    - Sb/GaAs reconstruction coexistence

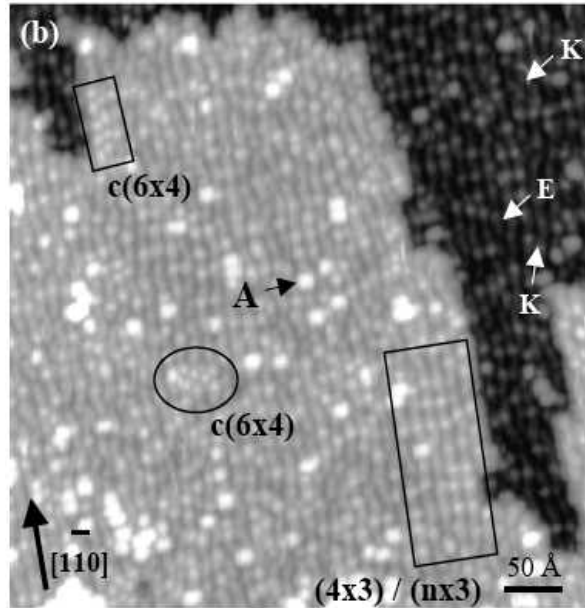


## Outline

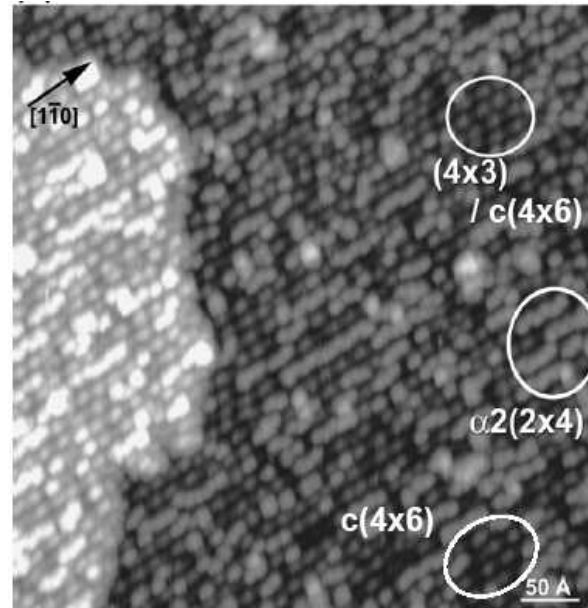
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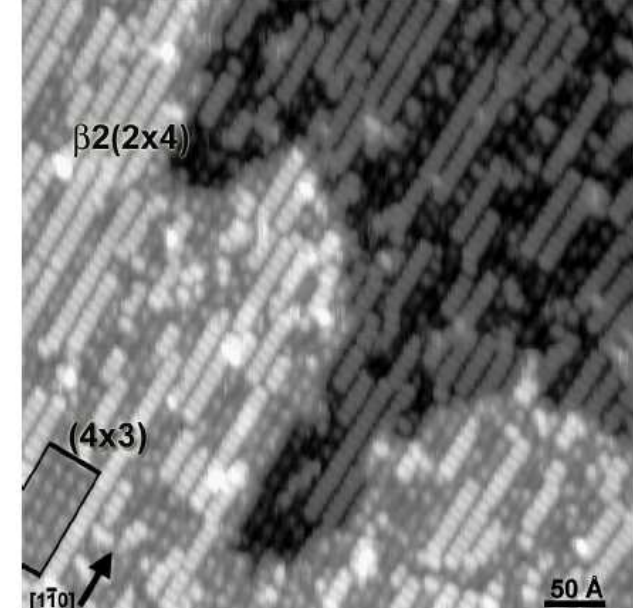
## Alloy Reconstructions Distinct from Constituents



- $\text{In}_{0.53}\text{Ga}_{0.47}\text{As}/\text{InP}$
- Lattice Matched
- (4x3) reconstruction



- $\text{In}_{0.27}\text{Ga}_{0.73}\text{As}/\text{GaAs}$
- 2% compressive Strain
- (4x3) and  $\alpha 2(2 \times 4)$  reconstructions



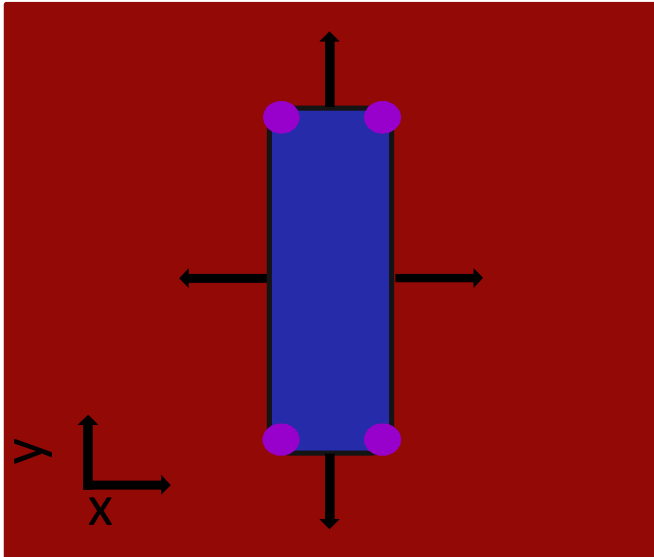
- $\text{In}_{0.81}\text{Ga}_{0.19}\text{As}/\text{InP}$
- 2% compressive Strain
- (4x3) and  $\beta 2(2 \times 4)$  reconstructions

# Elastic Relaxation Drives Coexistence of Reconstructions

- Strained InGaAs films exhibit nanoscale domains of different reconstructions
- Well behaved distribution of anisotropic domains

- Thermodynamic model for reconstruction coexistence\*

$$\Delta G = \underbrace{2(s+t)\beta}_{\text{Boundary Energy}} + \underbrace{st\Delta\gamma}_{\text{Surface Energy}} + \underbrace{U(s,t)}_{\text{Elastic Relaxation Energy}} + \underbrace{E_c}_{\text{Corner Energy}}$$

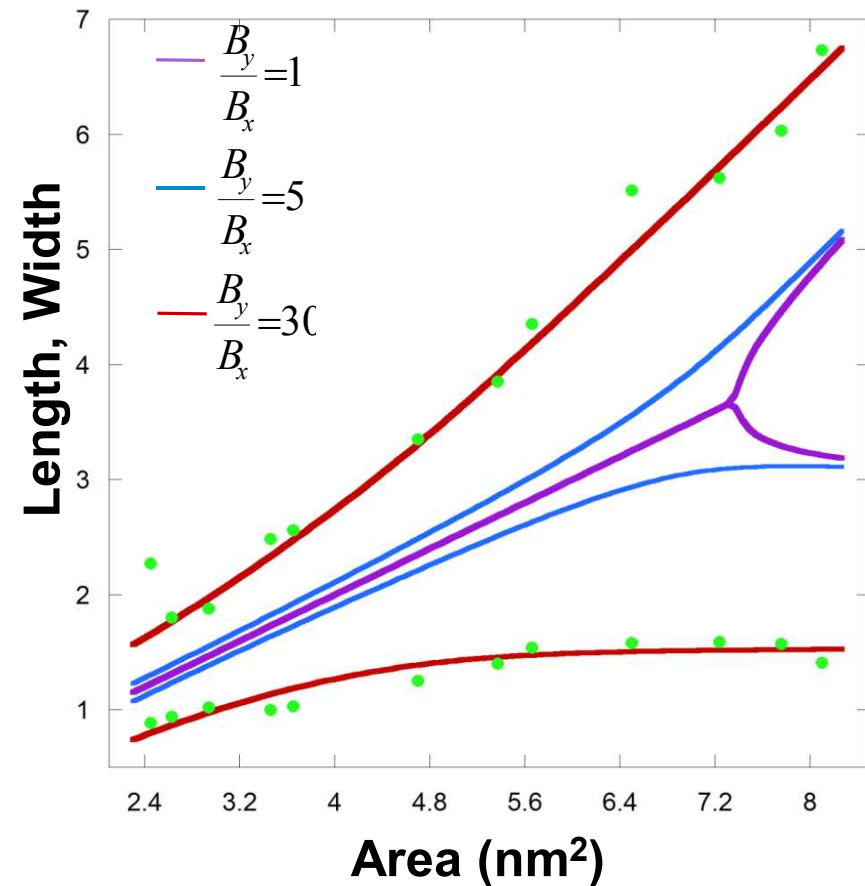


- Competing Interactions
  - Minimize surface energy
  - Maximize elastic relaxation
- Detailed evaluation of model requires knowledge of physical terms such as surface energy, surface tension, and boundary energy
  - Utilize DFT to determine values



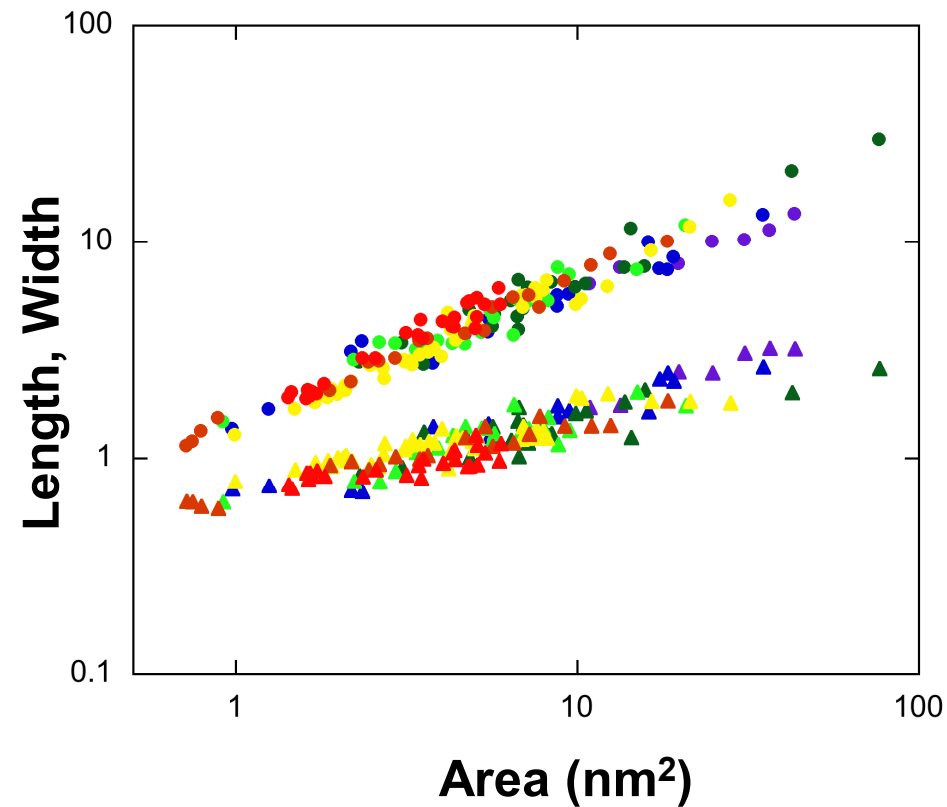
# Anisotropic Boundary Energy

- For an isotropic boundary energy  $\Delta G = 2(s + t)\beta + st\Delta\gamma + U(s, t) + E_c$  the length and width are equal below a critical area.
  - Above critical area rectangular domains are stable
- Square domains are not stable at any size when boundary energy is anisotropic.
- Experimental Data is Consistent!!!

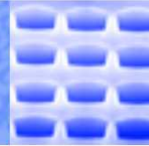


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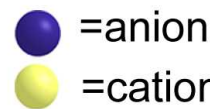
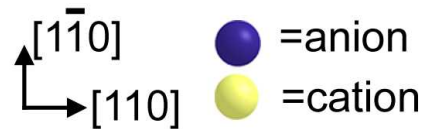
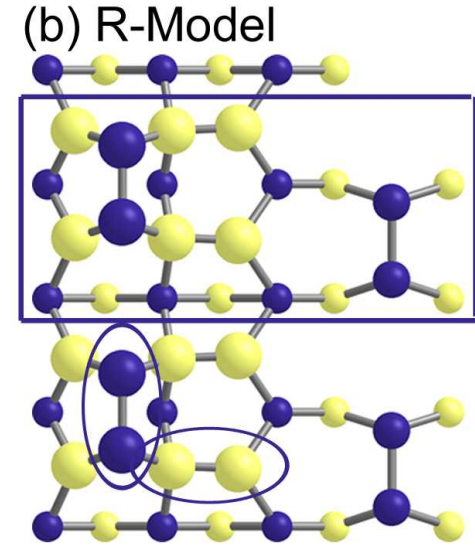
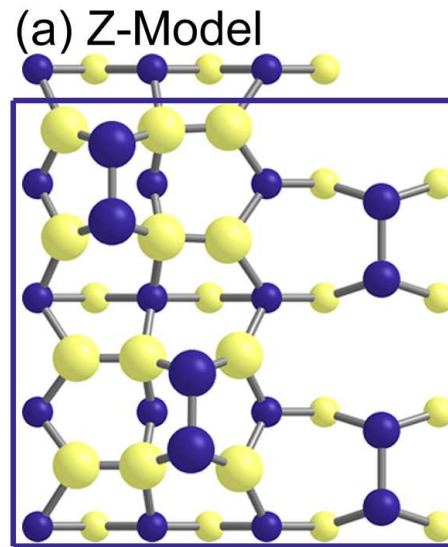
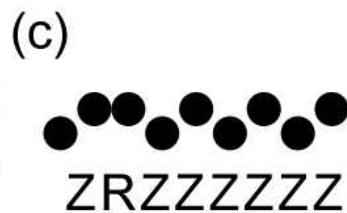
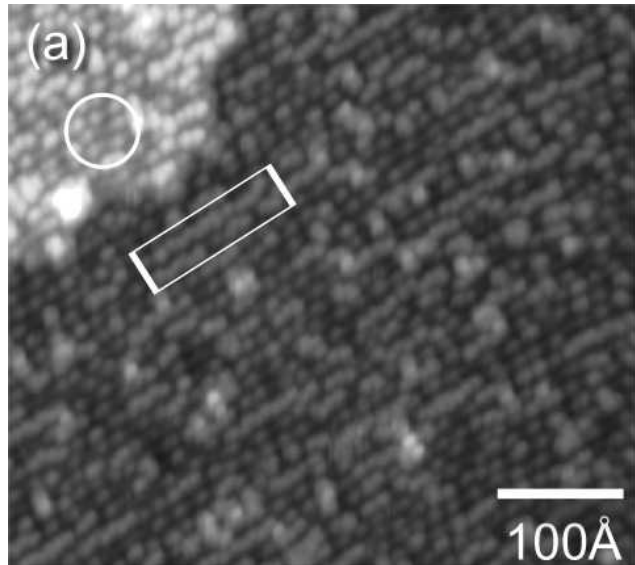


## Outline

Goal: Examine Atomic Surface Structure of III-V Alloys

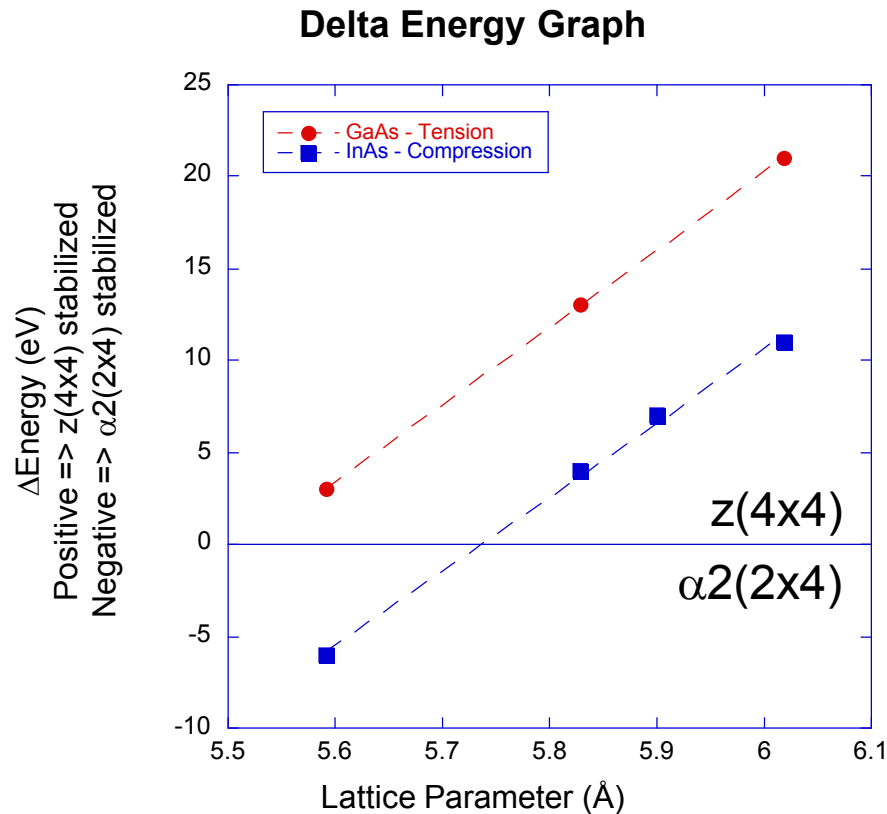
- Ternary alloys exhibit reconstruction *coexistence*
- Understanding the Role of Strain
  - Atomic size mismatch strain
    - $z(4 \times 4)$  reconstruction
    - $(4 \times 3)$  reconstruction (?)
  - Lattice mismatch strain
    - Sb/GaAs reconstruction coexistence

## The Alternating Dimer Configuration



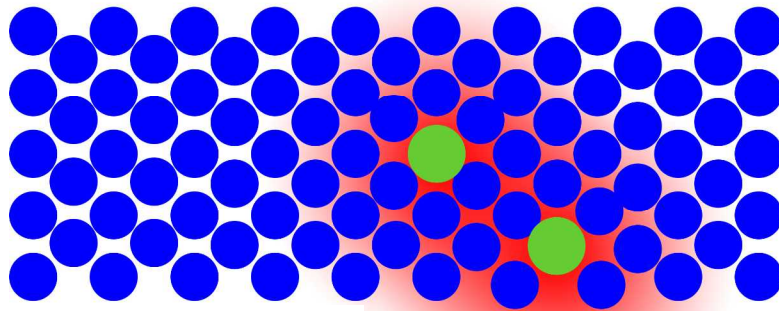
- $\text{In}_{0.27}\text{Ga}_{0.73}\text{As}$  experiment shows the surface dimer alternates position with an 80% incidence
  - z(4x4) reconstruction
- z(4x4) not seen in pure InAs or GaAs

# Misfit Strain Does Not Explain z(4x4)



- Misfit Strain is
 
$$\varepsilon = \frac{a_{film} - a_{substrate}}{a_{substrate}}$$
- Pure GaAs and InAs show slight z(4x4) stabilization
  - Not enough to overcome effects of temperature
- Misfit strain alone is not enough to stabilize z(4x4)

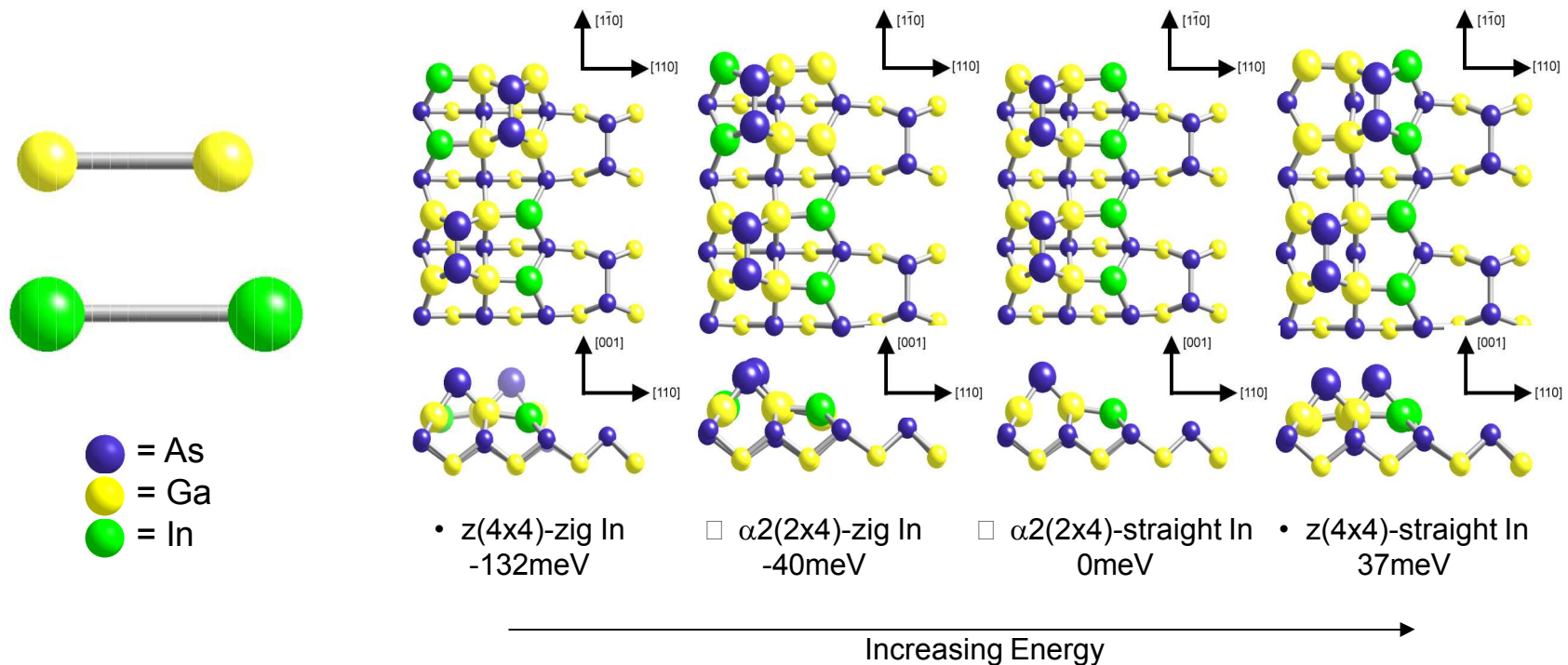
# Atomic Size Mismatch Strain



● = Element A

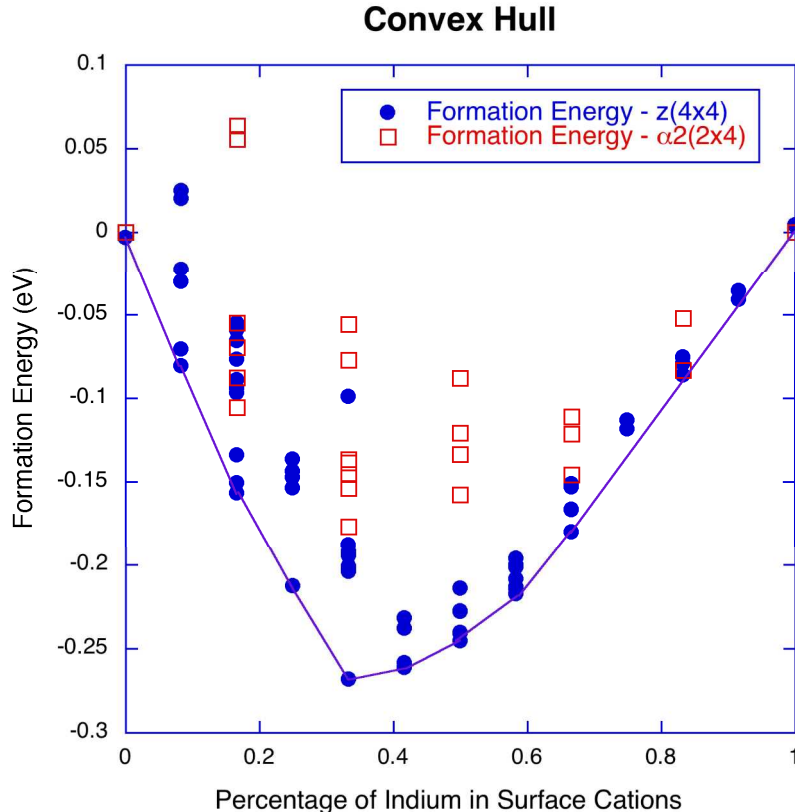
● = Element B

# Surface Alloying Stabilizes $z(4 \times 4)$



- Atomic size mismatch strain is induced by the size difference between two cations
- Placement of Indium in specific locations induces a surface solute strain and strongly stabilizes the  $z(4 \times 4)$ 
  - In atoms can relax the cation-cation bond
  - Large atoms can better relax in alternating pattern

# The Role of Atomic Size Mismatch Strain



- Convex hull displays formation energy of multiple configurations on a single graph
- z(4x4) is more stable than the α2(2x4) for most surface percentages of In
- Approximately 100meV stabilization for  $X_{\text{In}}=0.33$
- Atomic size mismatch strain stabilizes the surface reconstruction
- Suggests that the surface layer is most stable for a mixed composition

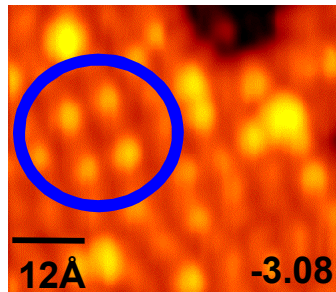


# (4x3) Surface Structure Still Unknown

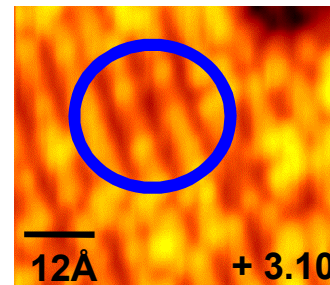
## Experimental STM of the (4x3) Structure

- $\text{In}_{0.53}\text{Ga}_{0.47}\text{As}$
- Annealed 25 min at growth conditions

Filled state image



Empty state image



Filled State Image

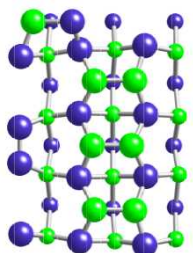
- Bright spots on dim rows

Empty State Image

- Bright spots and rows

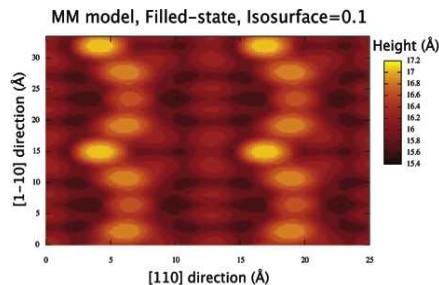
## DFT Simulated STM of the (4x3) Structure\*

Model

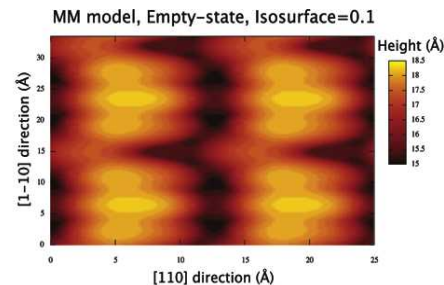


Millunchick et al

Filled state image



Empty state image



### Cation Dimer Model

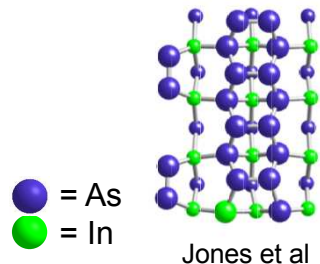
- Mixed Termination
- Anion Dimers
- Cation Dimers
- Hetero Dimers
- As coverage = 0.5ML

Millunchick et al. *Surf. Sci.* **550** 1 (2004)

### Anion Dimer Model

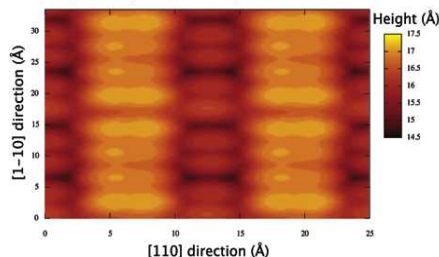
- Anion Termination
- Anion Dimers
- Excess Anion Dimer
- As coverage = 1.5ML

P.A. Bone et al., *Surf. Sci.* **600** 973 (2006)

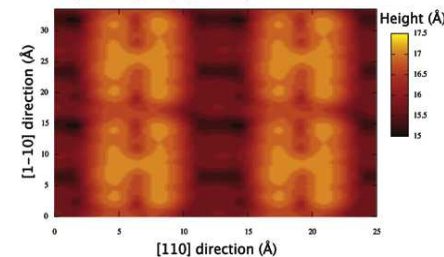


Jones et al

SSJ model, Filled-state, Isosurface=0.1

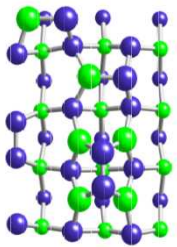


SSJ model, Empty-state, Isosurface=0.1

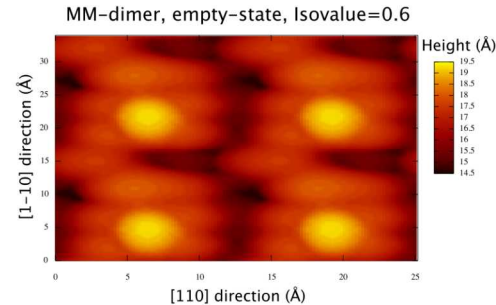
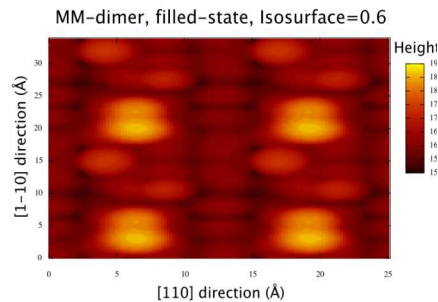


\*J. Tersoff, D.R. Hamann, *Phys. Rev. B.* **50** 1998 (1983)

# (4x3) Likely Stabilized By Solute Strain



Riposan

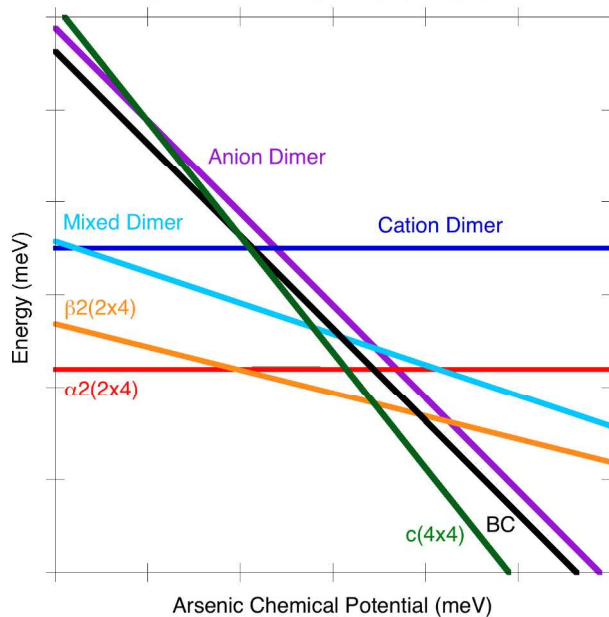


## Mixed Dimer Model

- Mixed Termination
- Anion Dimers
- Hetero Dimers
- As coverage = 0.83ML

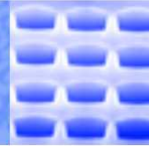
A. Riposan, (PhD dissertation)  
University of Michigan

## Surface Reconstructions at InP Lattice Parameter



Surface stability follows lowest energy line as a function of chemical potential

- Best experimental STM to DFT STM match is mixed dimer model
- Energy remains too high
- Misfit strain does not sufficiently lower energy
- Alloy induced solute strain may be the key

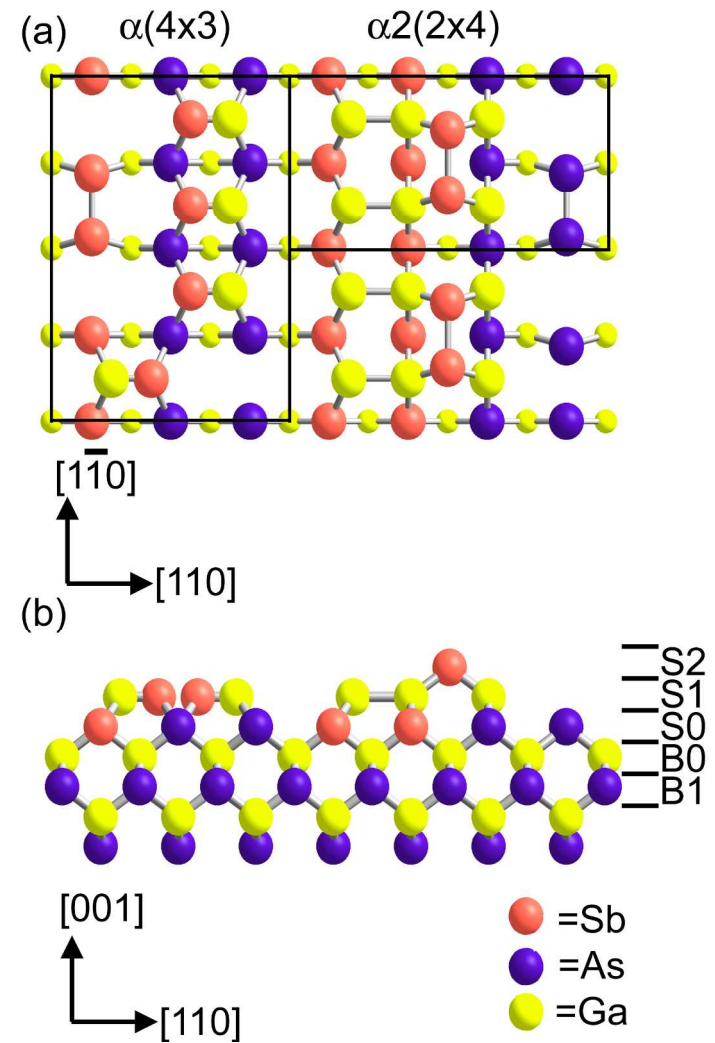
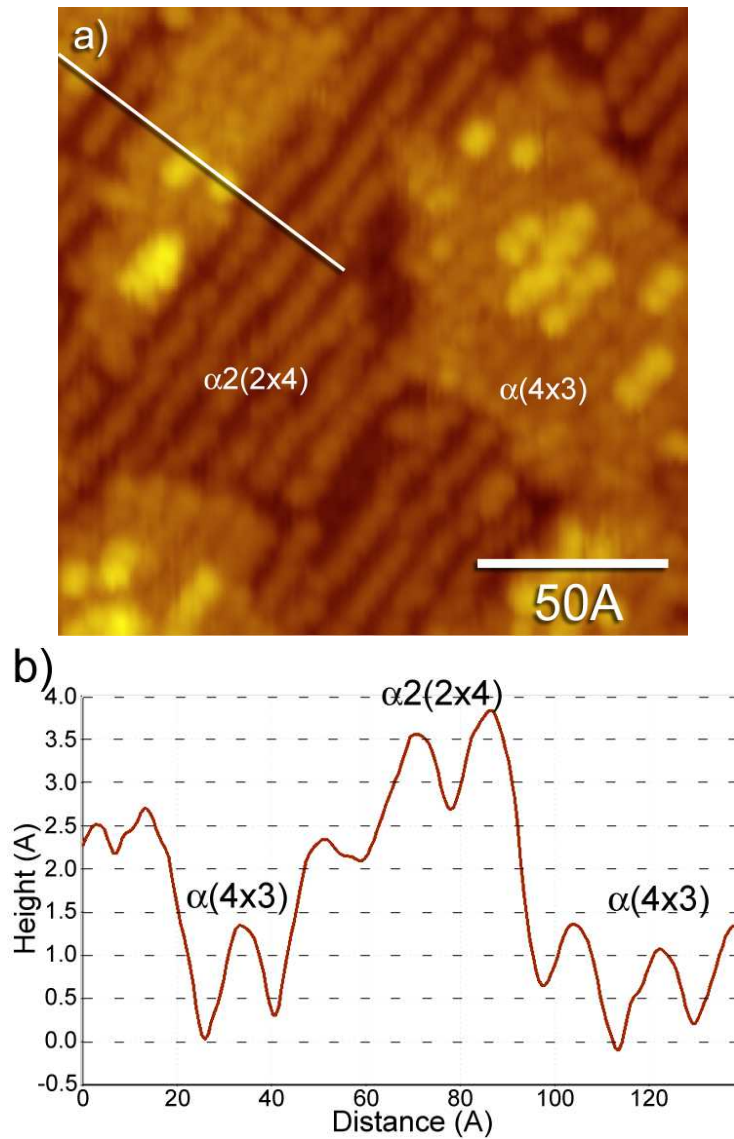


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    - Sb/GaAs reconstruction coexistence

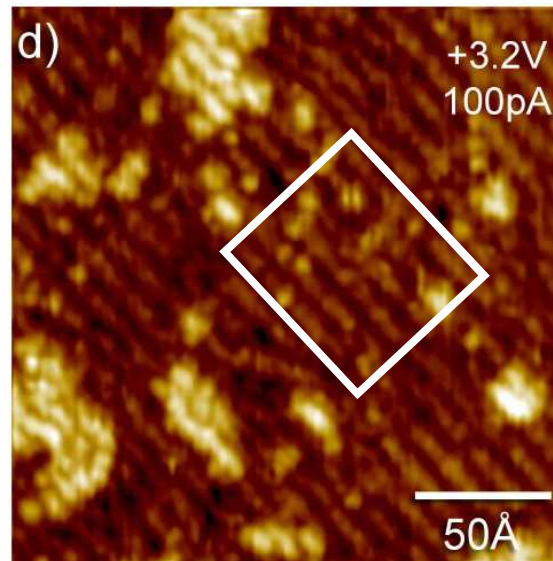
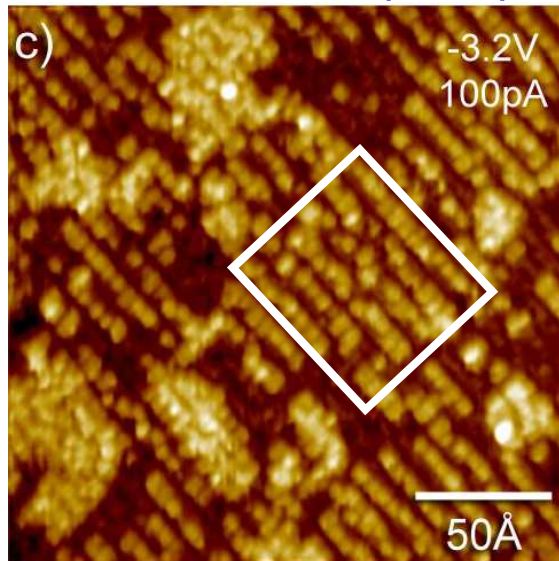
# Reconstructions on Sb/GaAs





# Surface Reconstruction of Sb/GaAs(001)

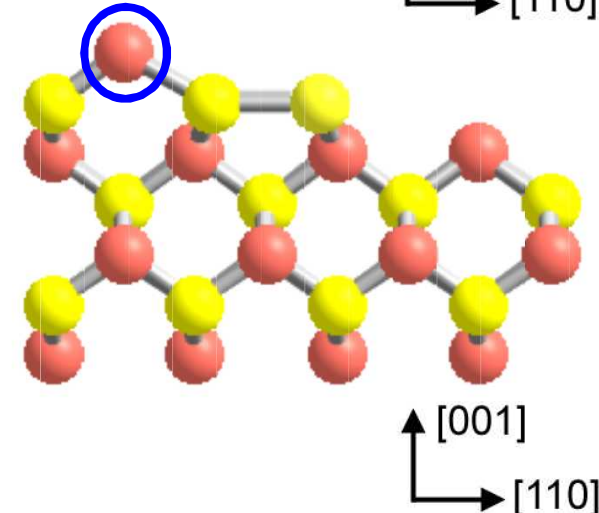
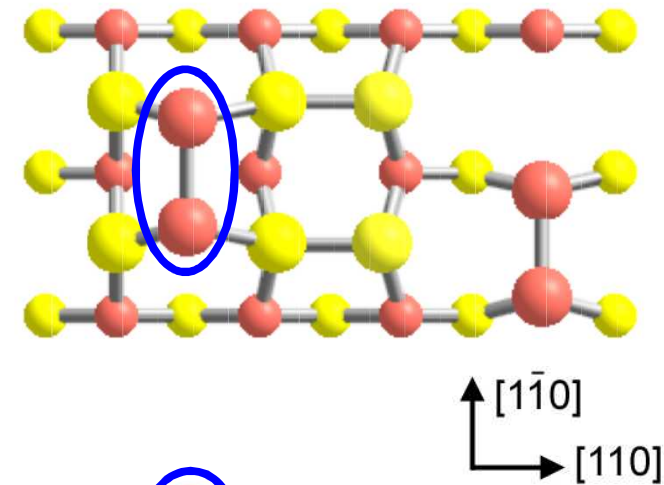
## $\alpha 2(2 \times 4)$ reconstruction



- $\alpha 2(2 \times 4)$ 
  - Common to GaAs
  - Not common to bulk GaSb

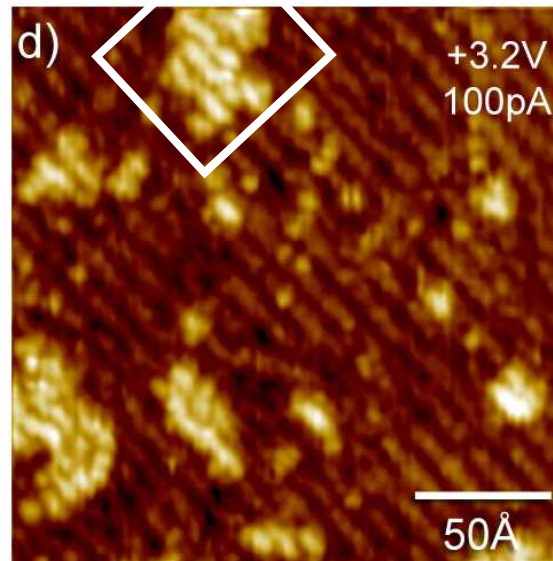
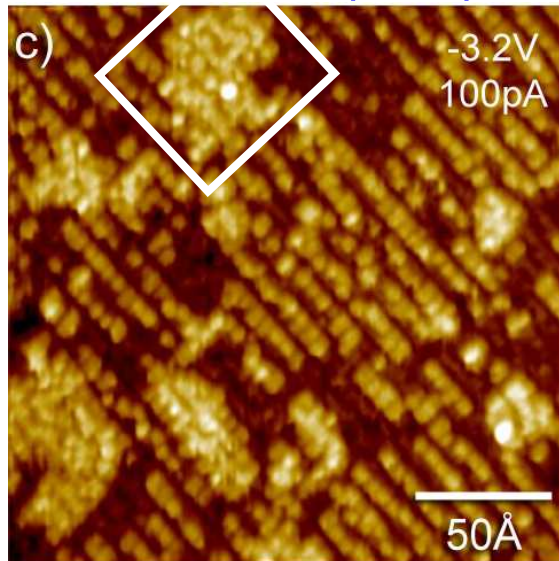
● = Ga  
● = Sb

Some atoms  
omitted for clarity



# Surface Reconstruction of Sb/GaAs(001)

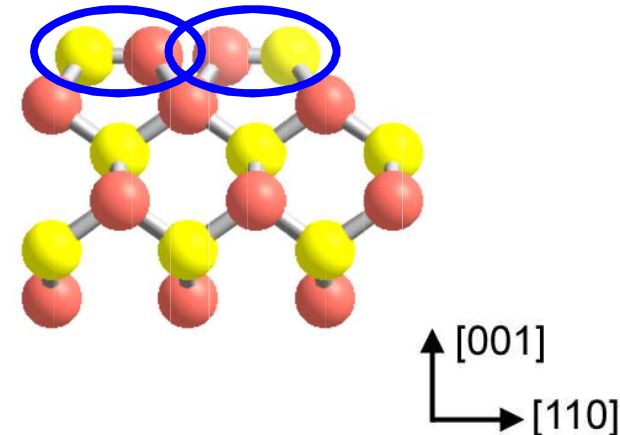
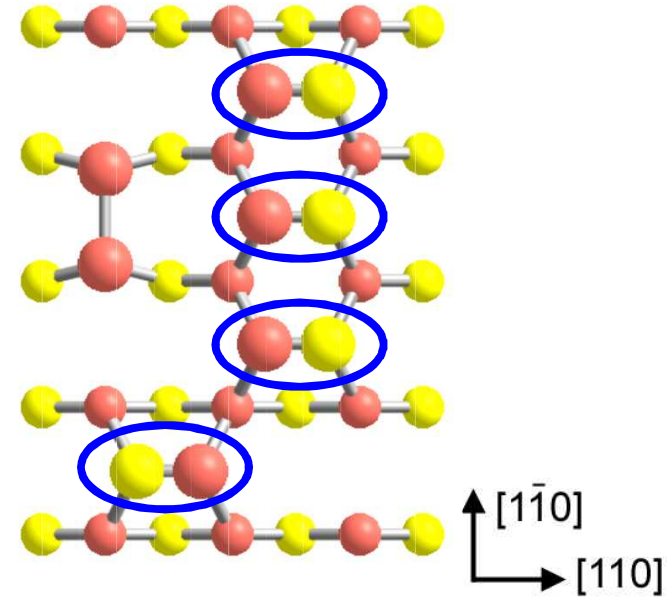
$\alpha(4 \times 3)$  reconstruction



- $\alpha(4 \times 3)$ 
  - Bulk GaSb reconstruction
  - Not common to GaAs

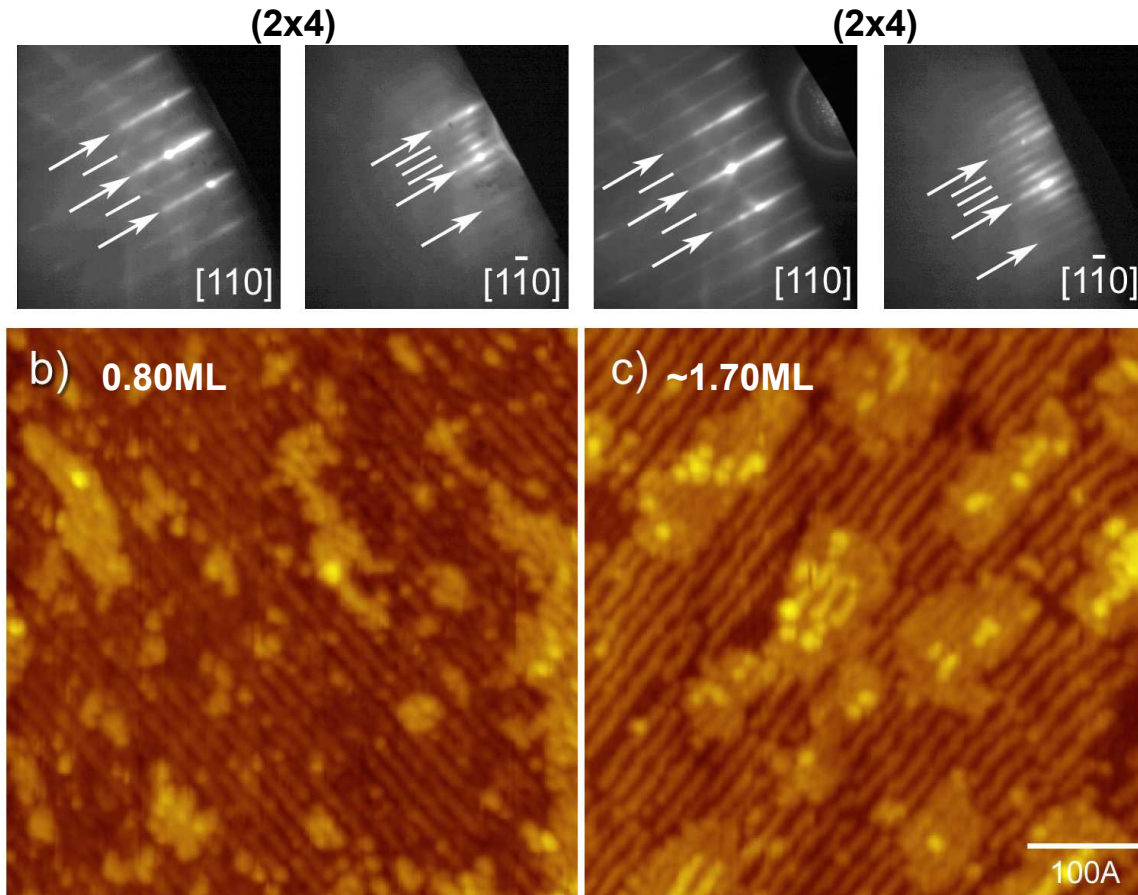
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*Some atoms omitted for clarity*





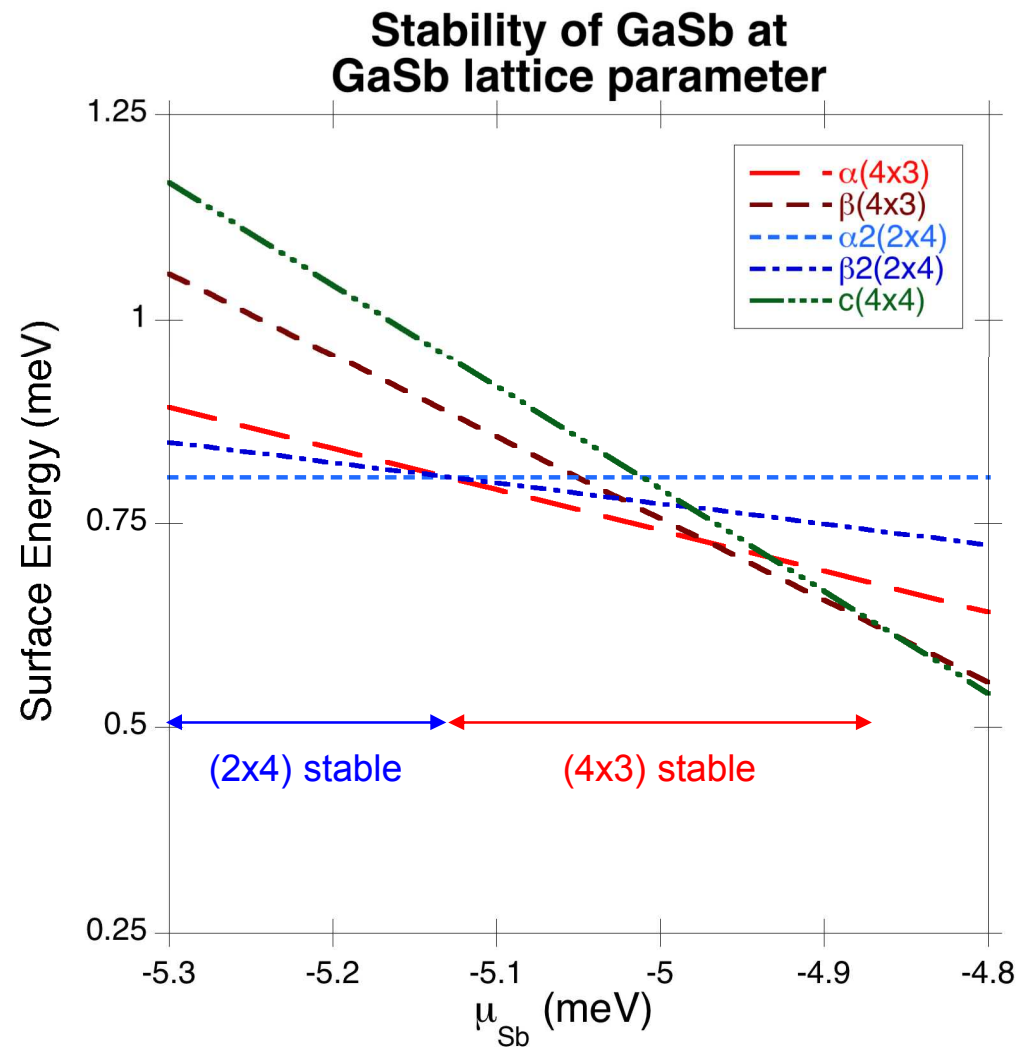
# Mixed Reconstruction Stability



RHEED of the samples after quenching (top), and room temperature STM images (bottom) for increasing thickness of Sb/GaAs. b) 0.8ML, c)  $\sim 1.7$ ML. The box in (a) indicates an area of x3 rows within the disordered surface.

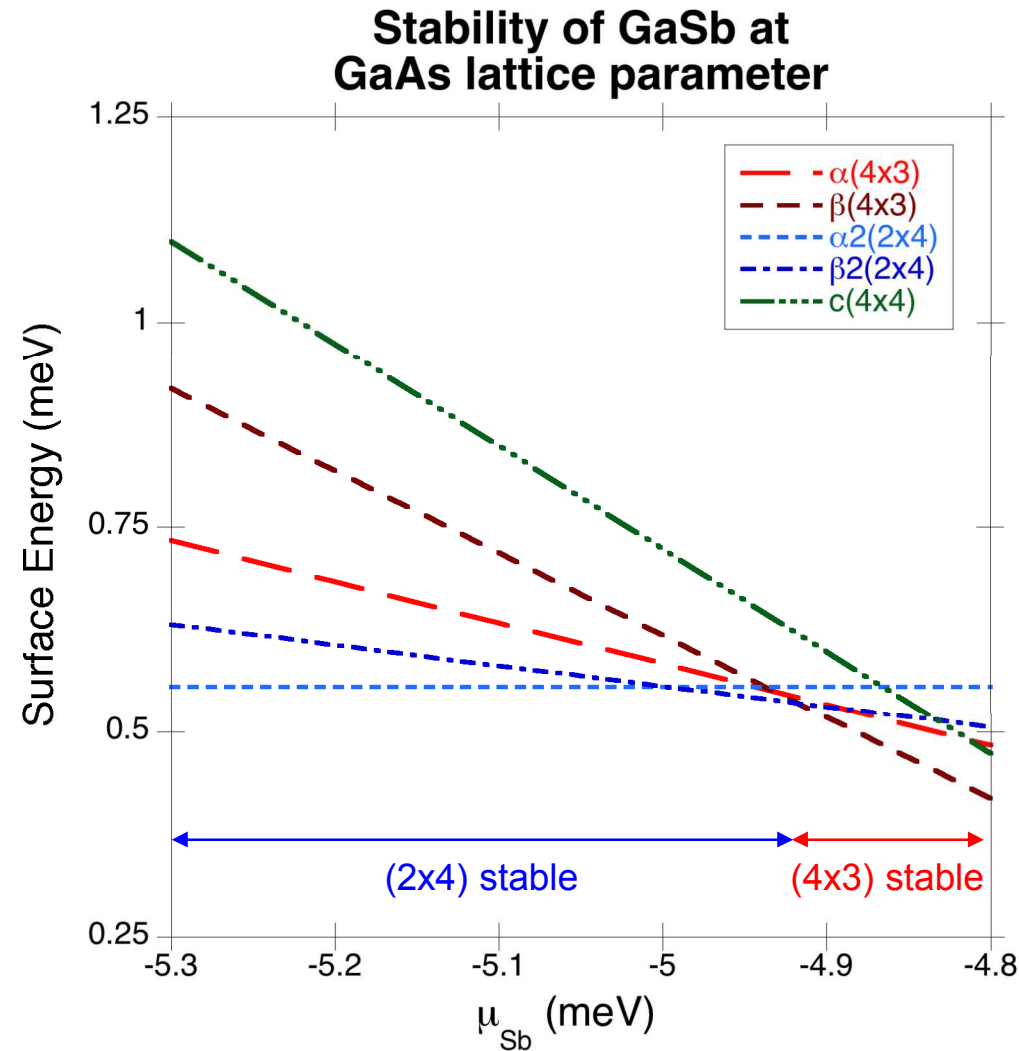
# Elastic Relaxation

- GaSb Lattice Parameter
  - Lowest line = stable reconstruction
  - (4x3) reconstruction stable for much of  $\mu_{\text{Sb}}$ 
    - $\mu_{\text{Sb}}$  = chemical potential of Sb
  - Stability with decreasing  $\mu_{\text{Sb}}$ 
    - c(4x4)
    - $\beta$ (4x3)
    - $\alpha$ (4x3)
    - $\alpha 2$ (2x4)



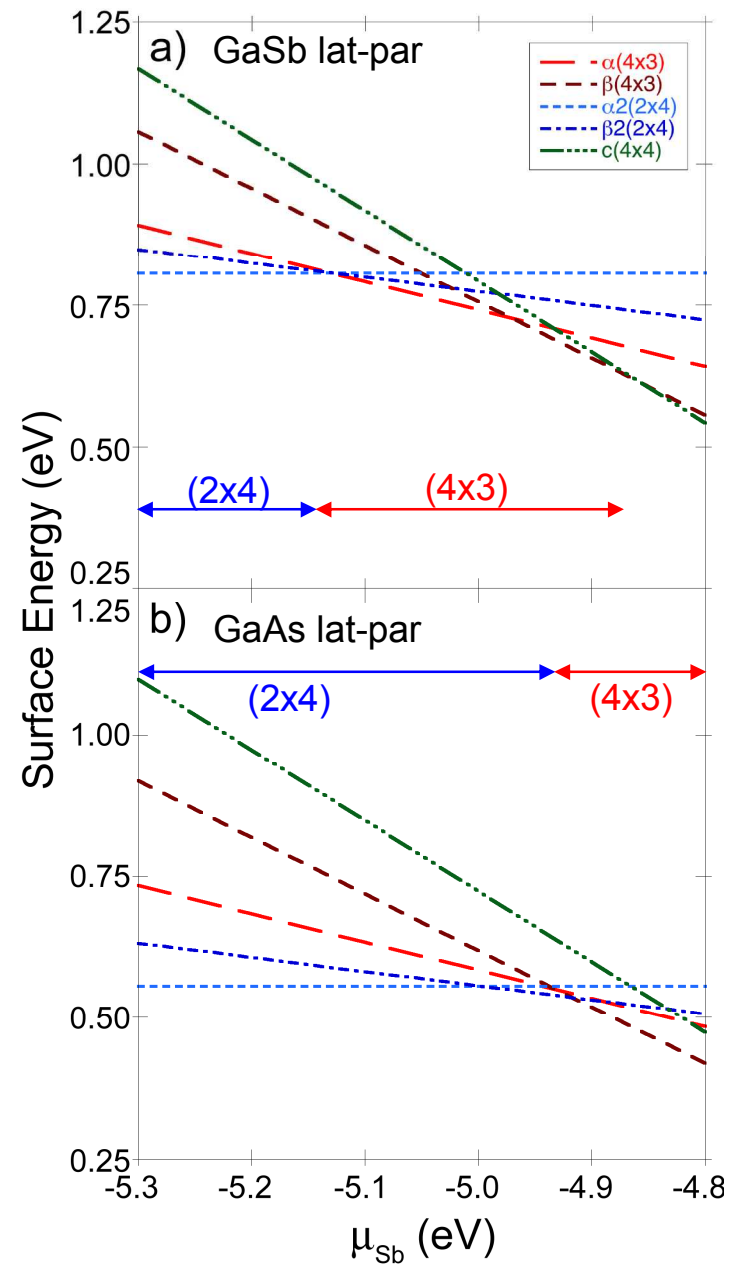
# Elastic Relaxation

- GaAs Lattice Parameter
  - Lowest line = stable reconstruction
  - (2x4) reconstruction stable for much of  $\mu_{\text{Sb}}$ 
    - $\mu_{\text{Sb}}$  = chemical potential of Sb
  - Stability with decreasing  $\mu_{\text{Sb}}$ 
    - $\beta(4\times 3)$
    - $\beta 2(2\times 4)$
    - $\alpha 2(2\times 4)$



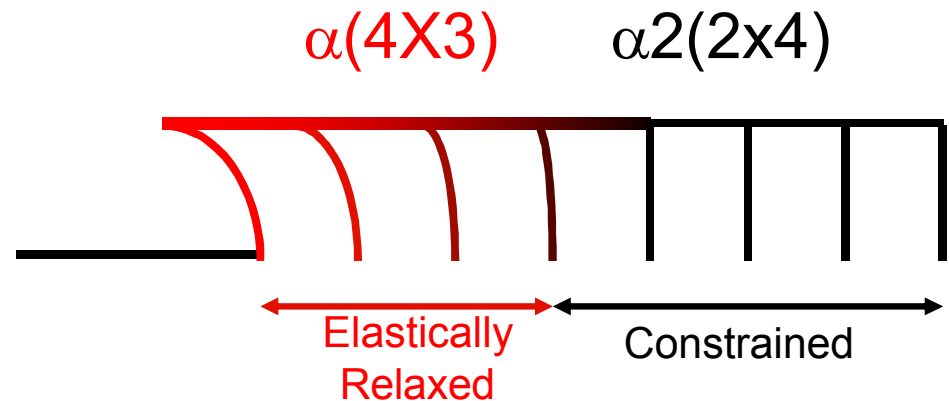
# Elastic Relaxation

- Mixed Reconstruction in GaSb/GaAs (001)
  - Mixed Anion System
- $\alpha 2(2 \times 4)$ 
  - Stable for smaller lattice parameters
  - Appears where lattice parameter is constrained
- $\alpha(4 \times 3)$ 
  - Stable at larger lattice parameters
  - Appears at terrace edges where lattice parameter can **elastically relax**



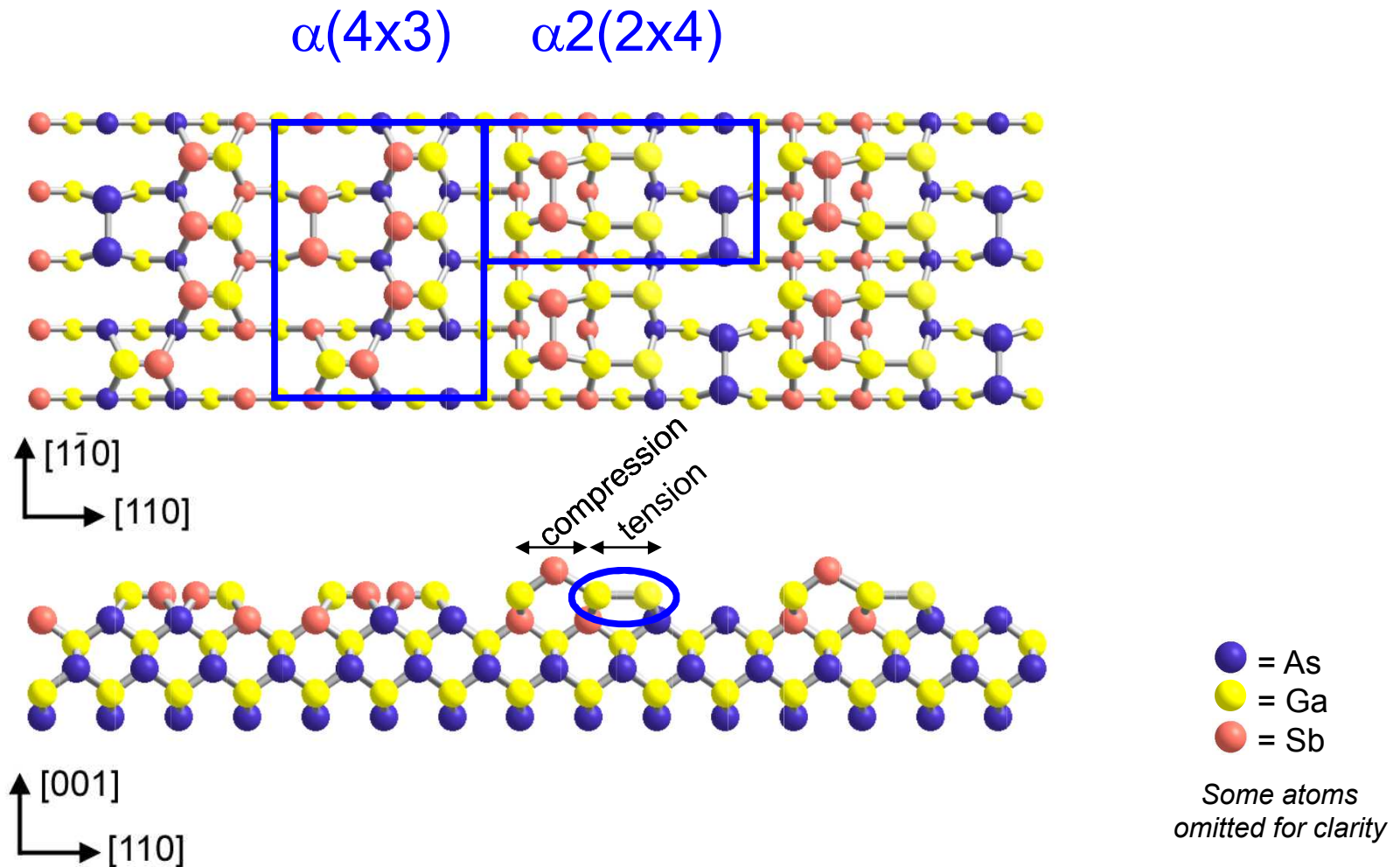
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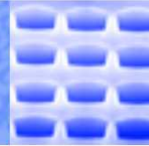




# $\alpha 2(2 \times 4)$ Relieves Compressive Strain



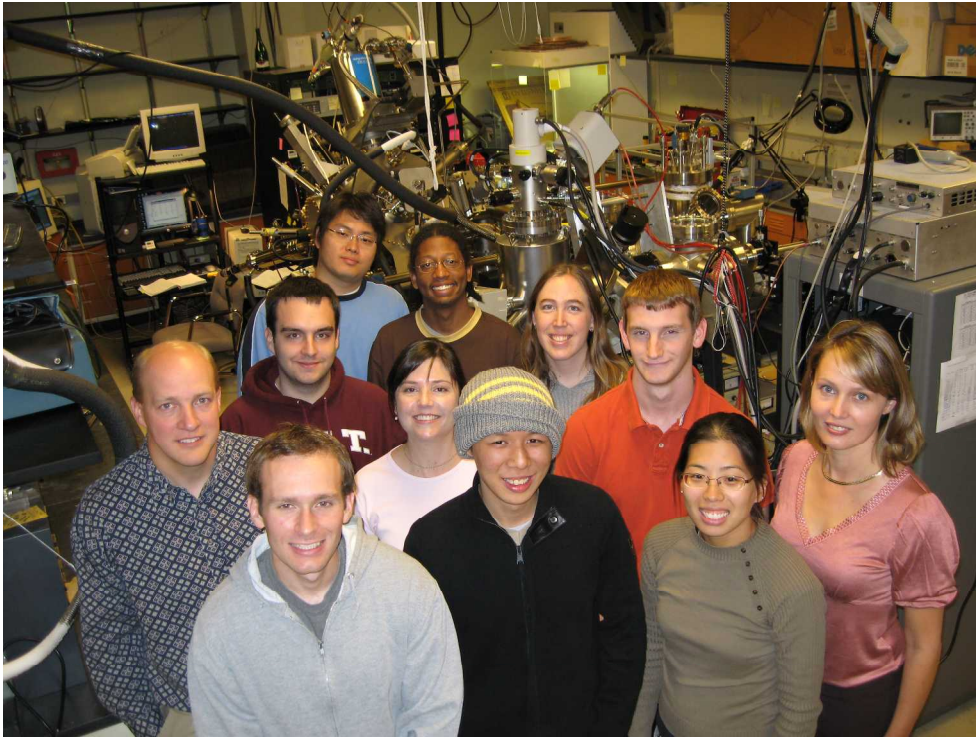




## Conclusions

- Ternary alloys are complex than their binary counterparts
- *Thermodynamic Coexistence Model*
  - Fit to data suggests that this is not a kinetic effect
- Role of Strain
  - Atomic Size Mismatch Strain
    - Stabilizes the  $z(4 \times 4)$  reconstruction
    - Mixed cation surface
    - Alloying necessary to DFT treatment
  - Lattice Mismatch Strain
    - Elastic relaxation at step edges => complex strain field on surface
    - May stabilize surface coexistence as for Sb/GaAs

## Acknowledgements



In Kyu Eu, Trey Sears  
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Chris Pearson, Shannon Brooks-Lehnert,  
Mark Noordhoek  
Kevin Grossklaus, Andy Deng, Jenny Lee,  
Joanna Mirecki Millunchick

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Nanotechnologies, at Los Alamos  
National Laboratory (Contract DE-  
AC52-06NA25396) and Sandia  
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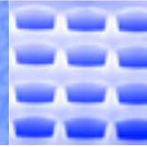
Other Collaborators:

[Anton VanDerVen](#)

University of Michigan

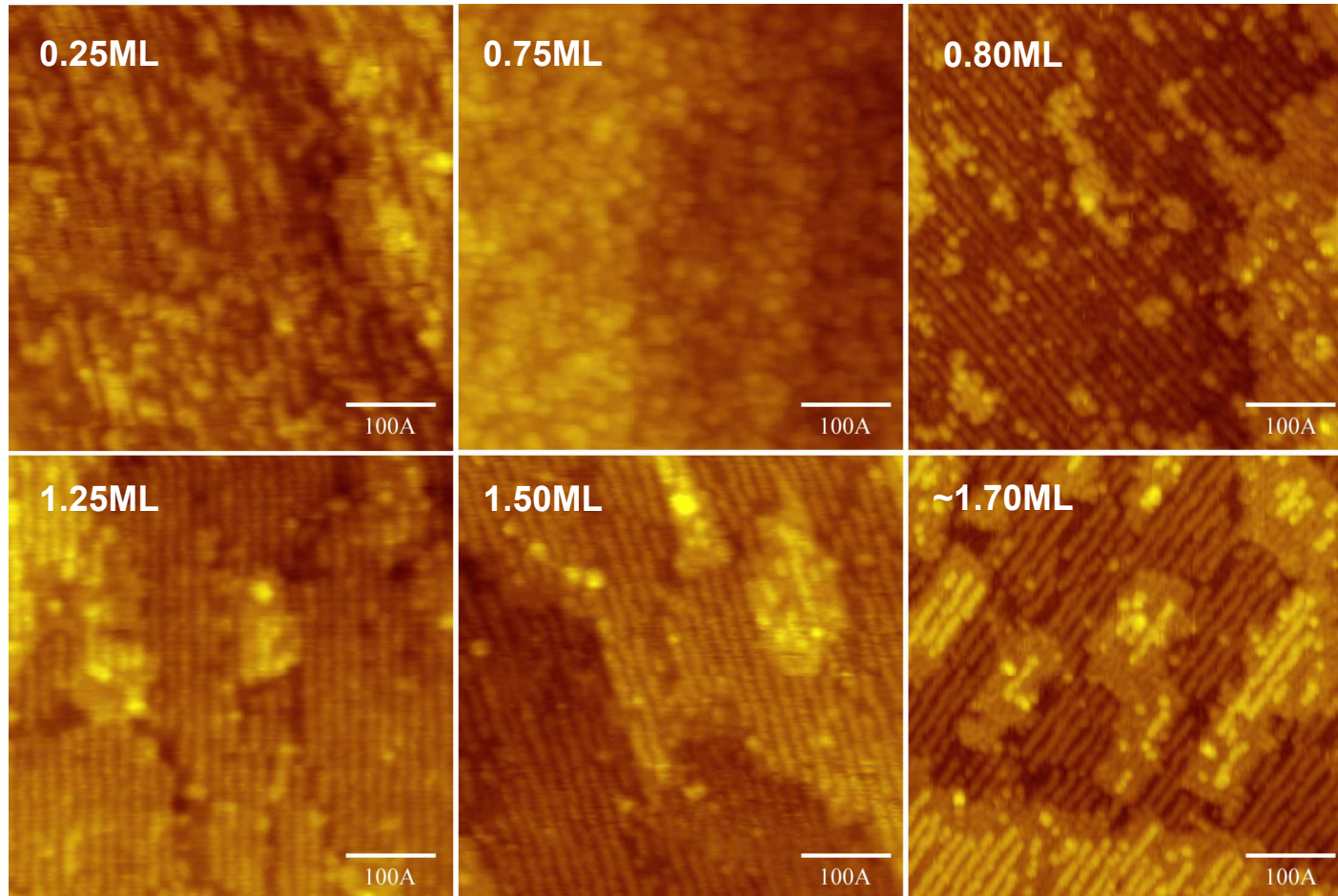
[Normand Modine](#)

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## Supplemental Slides

## Thickness Study



Thin layers of Sb/GaAs. Top left to right: 0.25ML, 0.75ML, 0.8ML.  
Bottom left to right: 1.25ML, 1.5ML, ~1.7ML



# Elastic Relaxation

$$U(s,t) = 2F_0 \left\{ \begin{array}{l} -s(1-\nu) \left[ \ln \frac{s}{a} - 1 \right] + u_1(a) \\ -t(1-\nu) \left[ \ln \frac{t}{a} - 1 \right] + u_1(a) \\ + 2\nu \left( s + t - \sqrt{s^2 + t^2} \right) + u_2(a) \\ + (1-2\nu) \left( s - \sqrt{s^2 + t^2} \right) + t(1-\nu) \ln \frac{\sqrt{s^2 + t^2} + t}{s^2 + t^2 - t} \\ + (1-2\nu) \left( t - \sqrt{s^2 + t^2} \right) + s(1-\nu) \ln \frac{\sqrt{s^2 + t^2} + s}{s^2 + t^2 - s} \end{array} \right\}$$

*Energy of Border s*

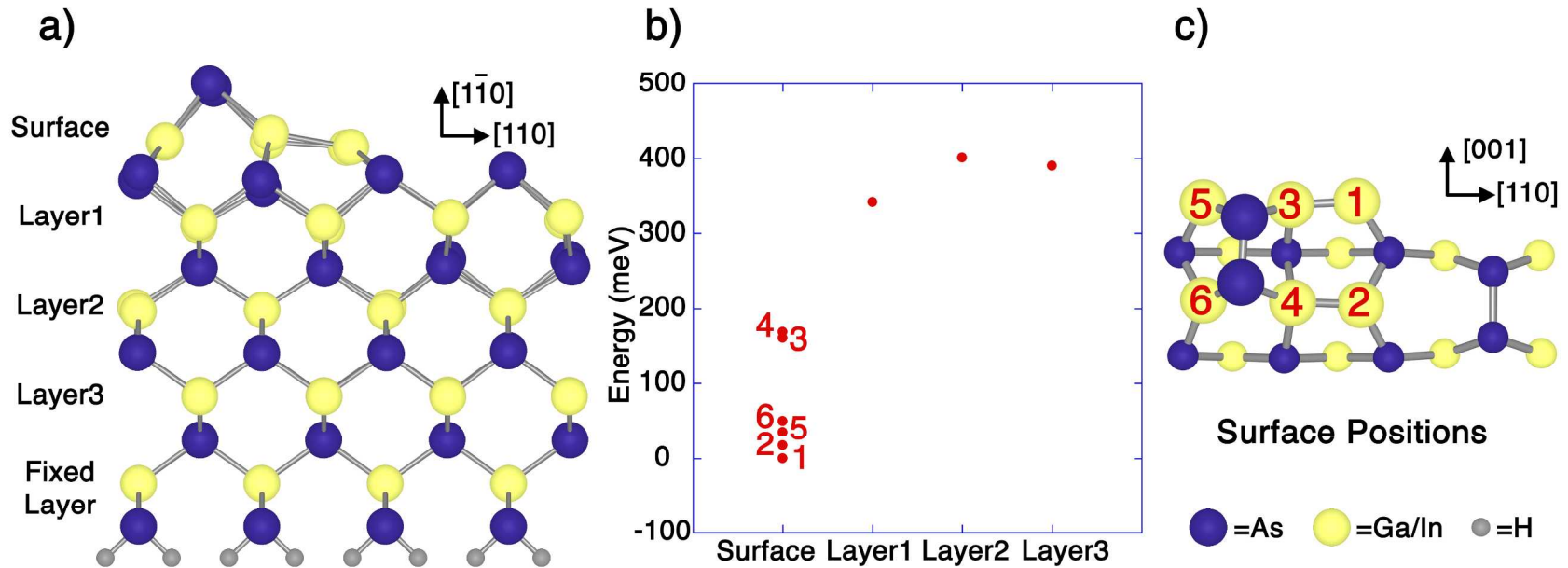
*Energy of Border t*

*Energy of Interaction where s,t meet*

*Interaction between parallel segments in s*

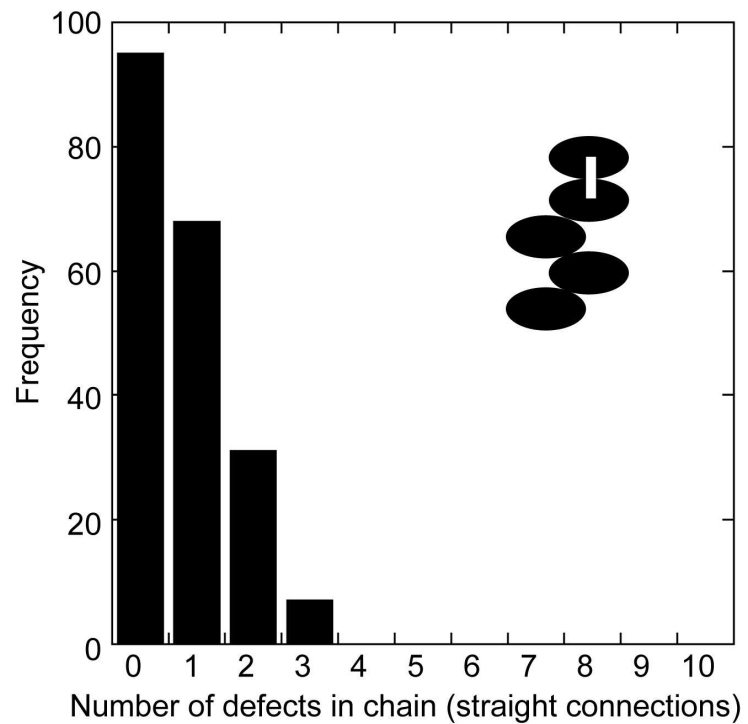
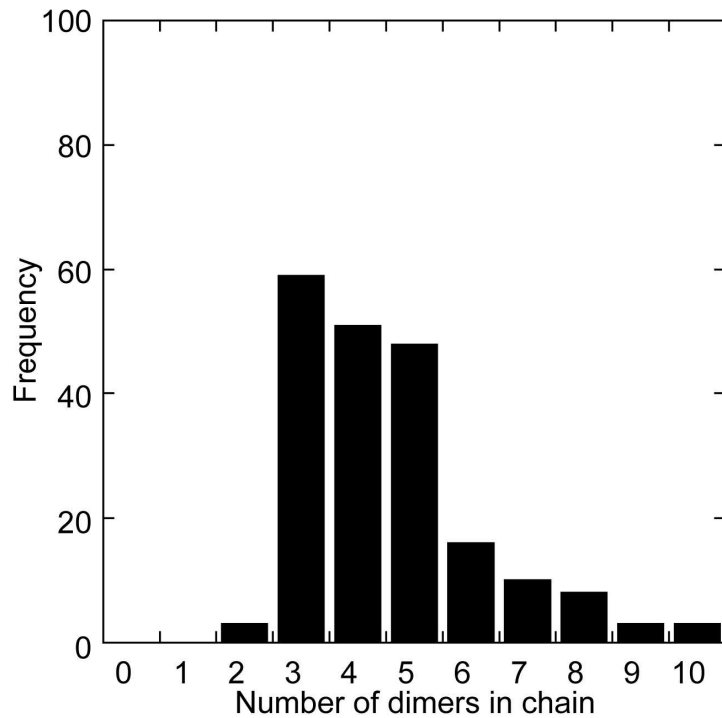
*Interaction between parallel segments in t*

# Indium Surface Segregation

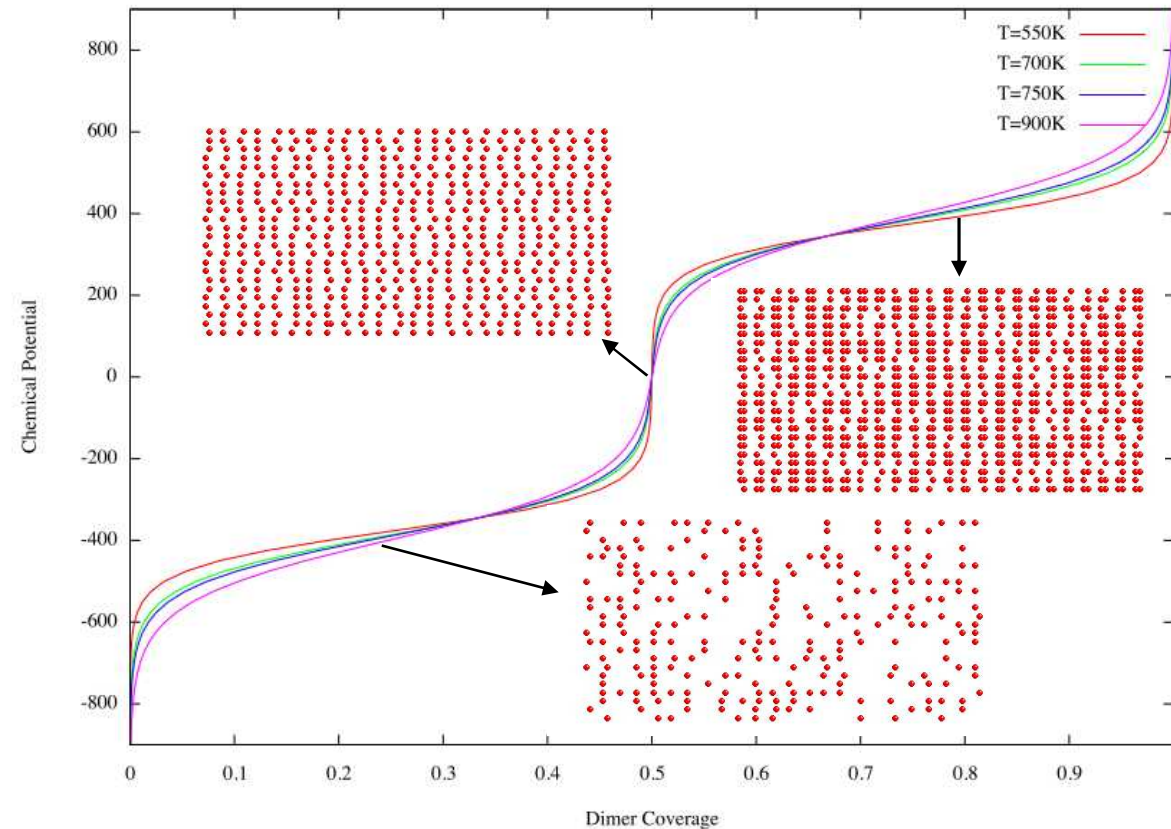




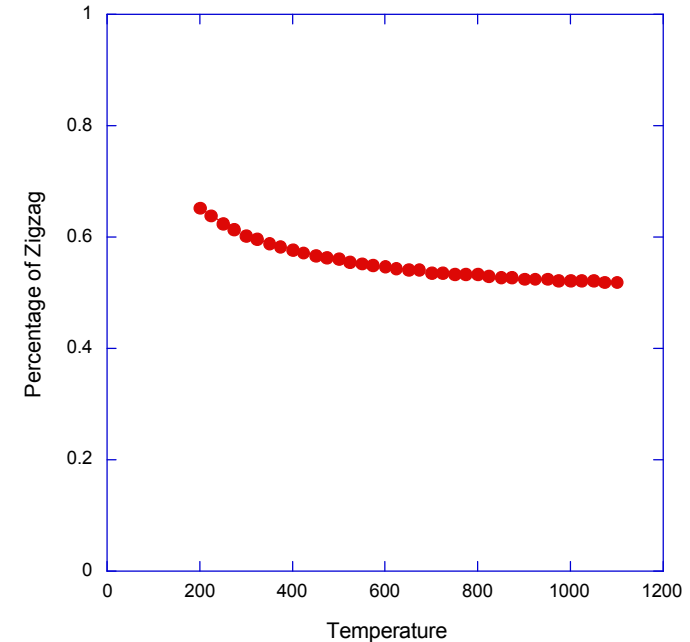
## $z(4 \times 4)$ Statistics



# Monte Carlo of Pure InAs



Percentage of Zigzag for 50% Dimer Coverage



- Large Stable areas for 50% dimer coverage
- Do see a slight favoring of the  $z(4 \times 4)$ , particularly for lower temperatures

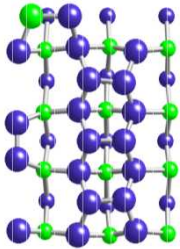
# Reconstructions in (4x3) Energy Comparison

Model

Filled state image

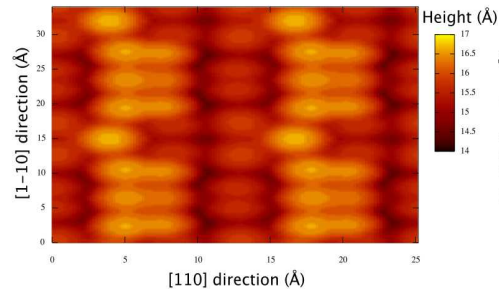
Empty state image

Model Characteristics

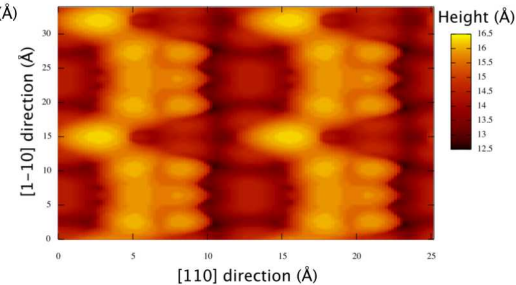


Barvosa-Carter et al  
*W. Barvosa-Carter, et al, Phys. Rev. Lett. 84 4649 (2000)*

Binary BC, filled-state, Isosurface=0.6

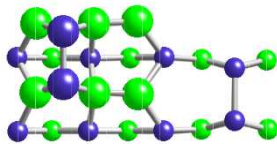


Binary BC, empty-state, Isosurface=0.6



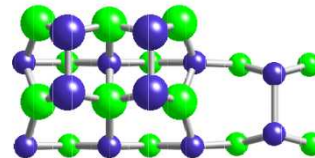
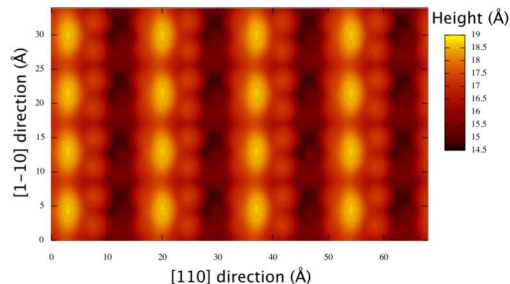
BC (4x3) model

- Mixed Termination
- Anion Dimers
- Hetero Dimers
- As coverage = 1.5ML

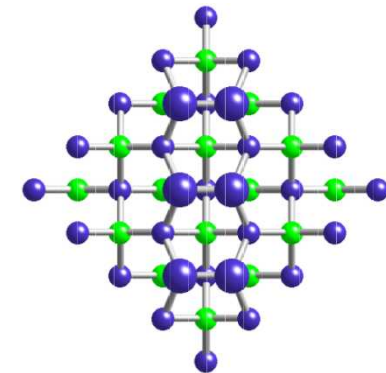
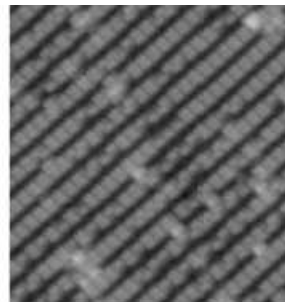


$\alpha_2(2 \times 4)$

Alpha2(2x4), Filled State, I=0.1



$\beta_2(2 \times 4)$



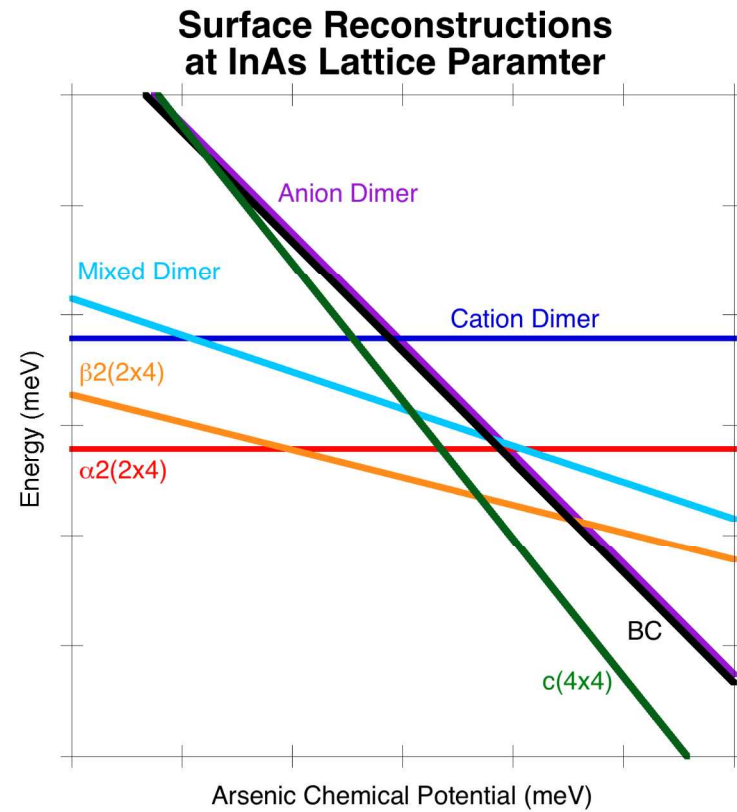
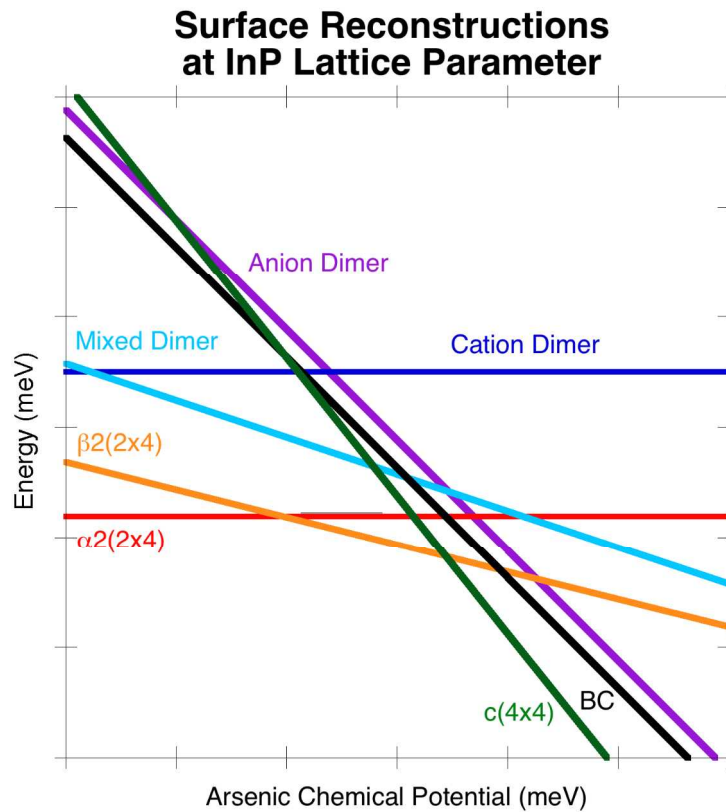
QuickTime™ and a  
(LZW) decompressor  
needed to see this picture.

● = As  
● = In

*Barvosa-Carter et al., Surface Science 499, L129 (2002).*

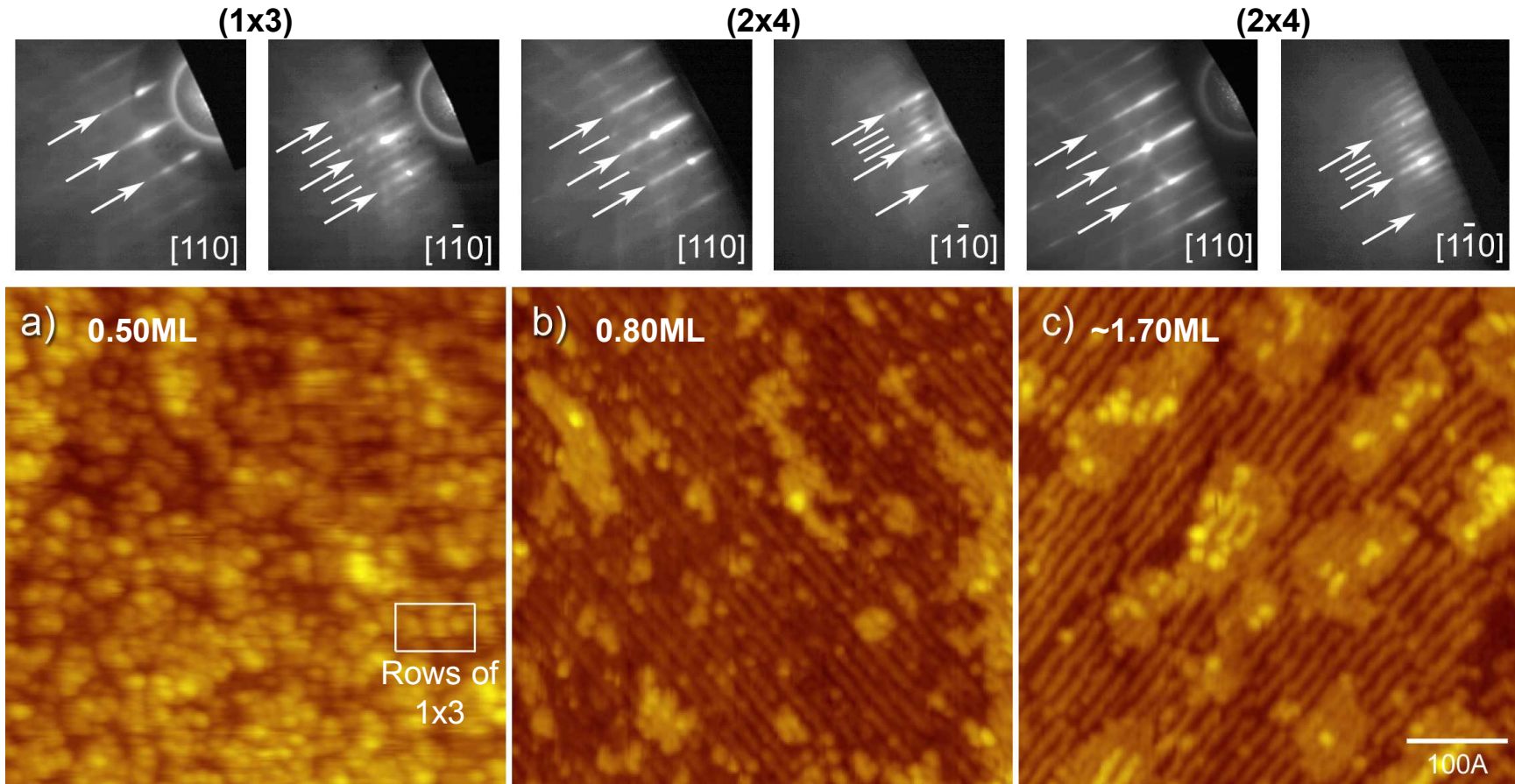
*LaBella et al., Surface Science 60, 1 (2005).*

# Effect of Misfit Strain on (4x3) Stability



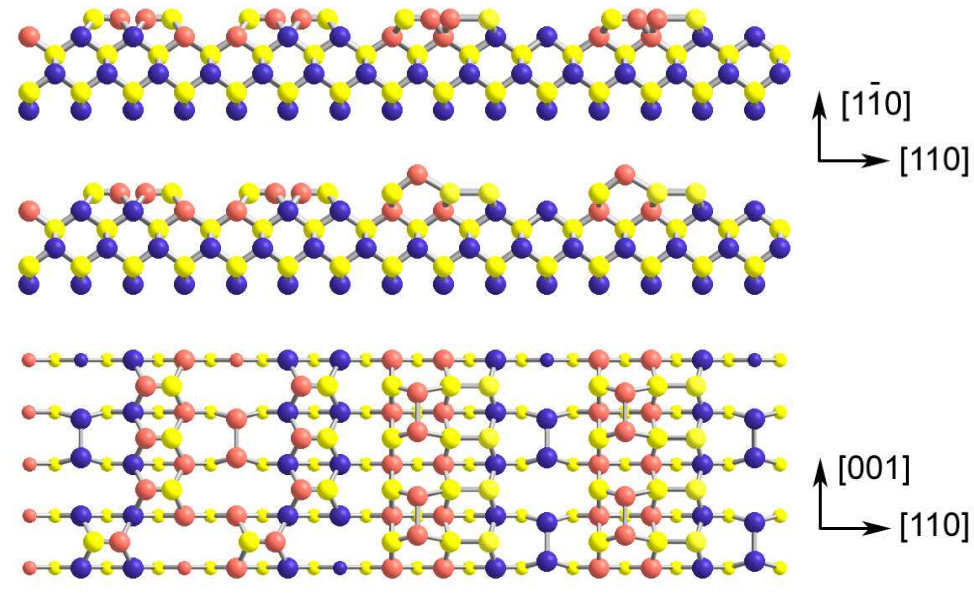
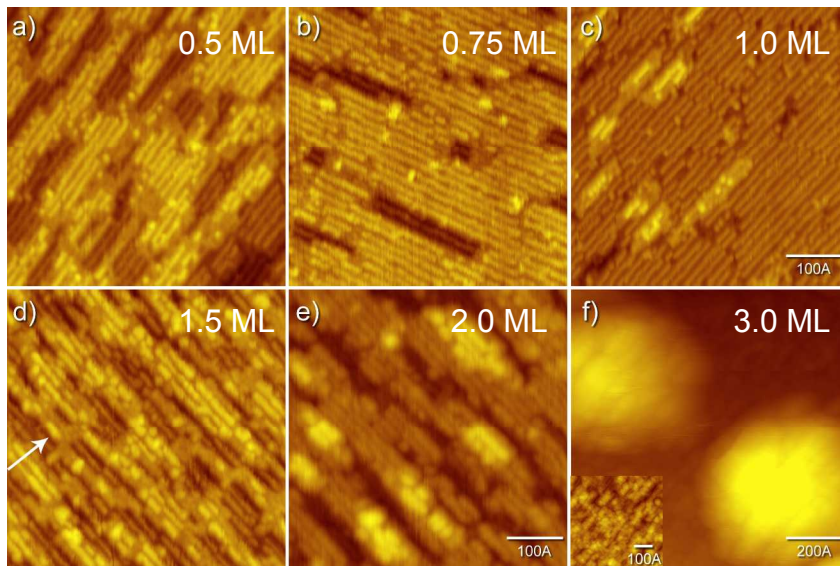
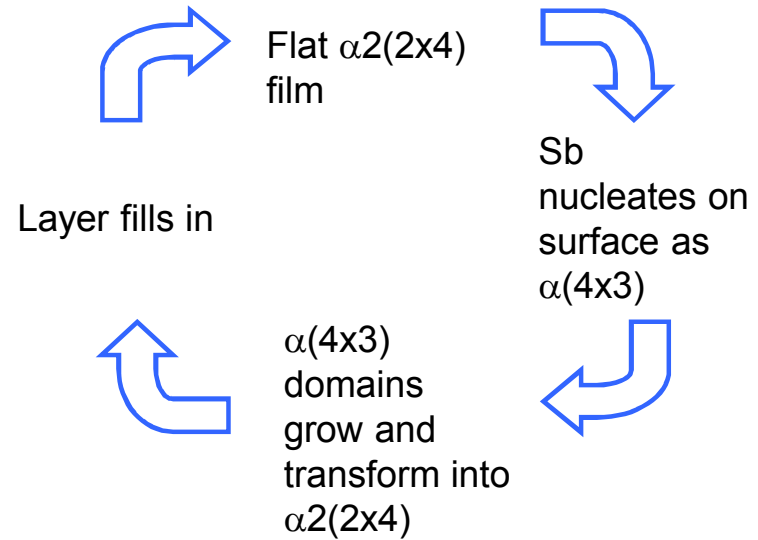


# Thickness Study



RHEED of the samples after quenching (top), and room temperature STM images (bottom) for increasing thickness of Sb/GaAs. a) 0.5ML, b) 0.8ML, c) ~1.7ML. The box in (a) indicates an area of x3 rows within the disordered surface.

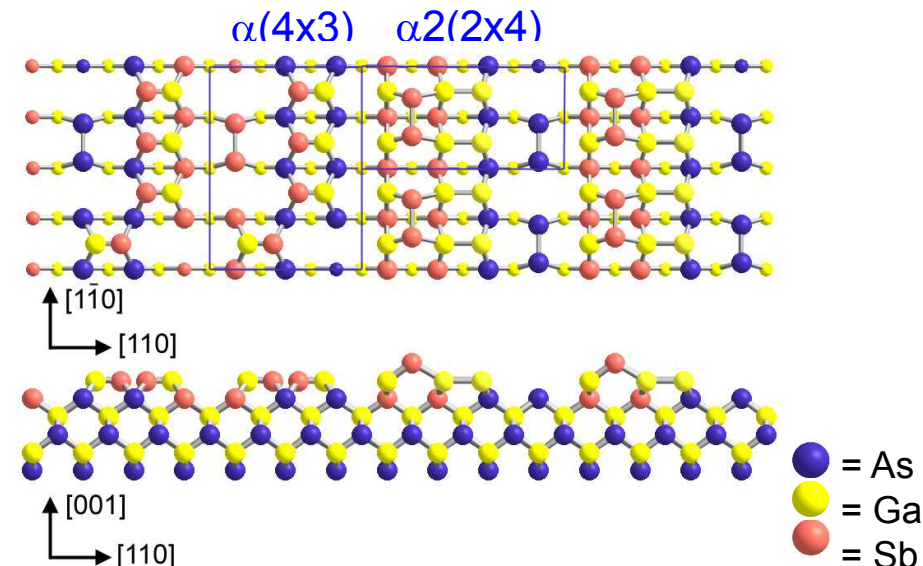
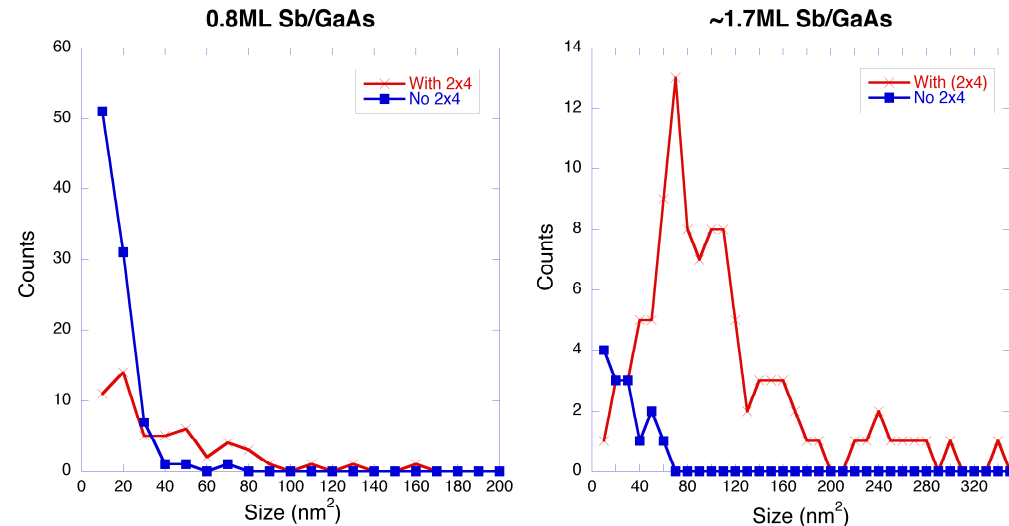
# Growth Model





# Critical Terrace Size

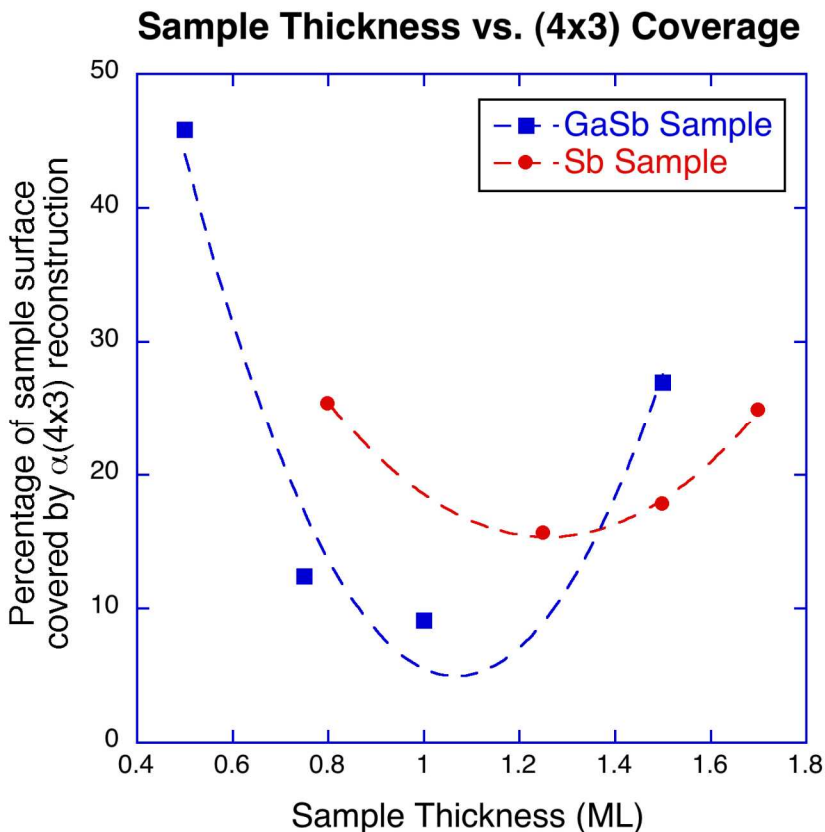
- Two types of terraces
  - $\alpha(4\times3)$  only
  - $\alpha(4\times3)$  at edge with  $\alpha_2(2\times4)$  in center
- Maximal size of  $\alpha(4\times3)$  terrace =  $30\pm 10\text{nm}^2$
- Suggests strain a factor in surface reconstruction stabilization
- The reconstructions are able to coexist next to each other
- Both have trench dimers which could help coexistence
- Notice the height difference



## Role of Ga

The percentage of the sample surface covered by either reconstruction changes during growth due not only to strain but also to the availability of Ga.

- (2x4) requires 9 Ga per unit area
- (4x3) requires 4 Ga per unit area



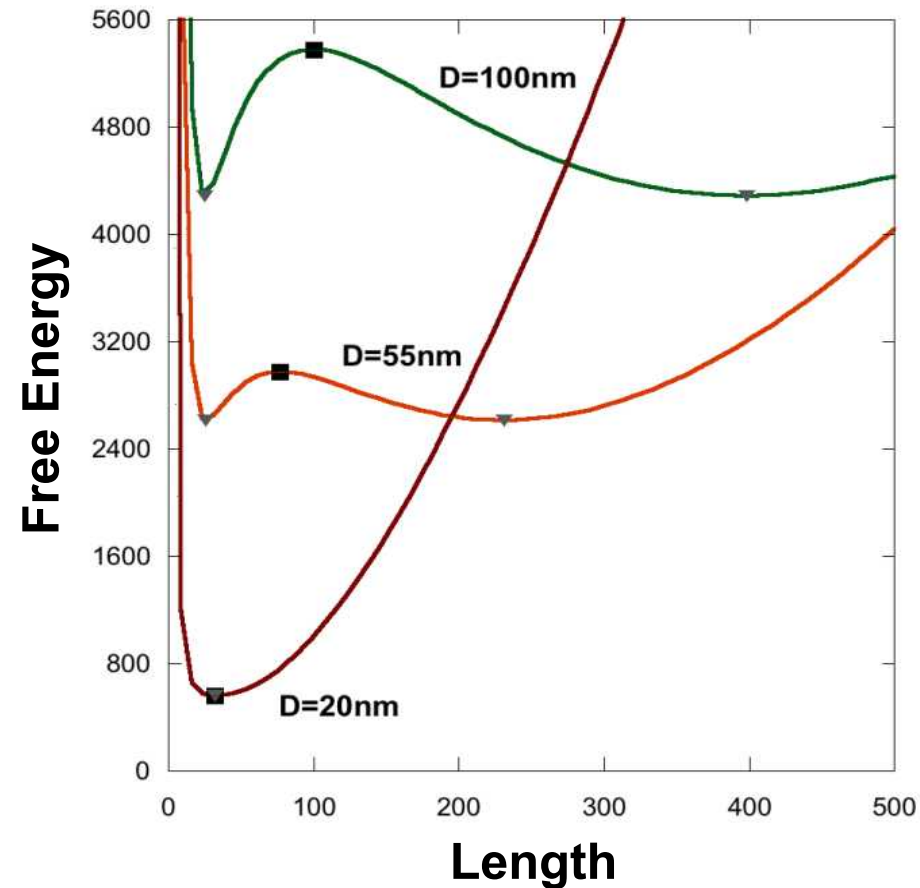
### GaSb growth

- Ga is present in the beam
- **Transformation is easy**
- Percentage of (4x3) on the surface varies dramatically
- Minimum percentage of Sb is ~1.05ML

### Sb Growth

- Ga is present only via diffusion from step edges
- **Transformation Ga-limited**
- Percentage of (4x3) is much more constant
- Thickness at which transformation complete is higher, ~1.25ML

- For small values of  $F$  the surface energy and boundary energy terms dominate
  - Above a critical size square domains are not energetically favored
  - Multiple rectangular shaped domains coexist
- But domains want to shrink to zero (i.e. disappear)



## High Unit Strain Energy

- For large values of  $F$ 
  - Still a critical size above which square domains are not energetically favorable
  - Domains are now globally stable.
- The equilibrium reconstruction domain shape is a quantum wire.

