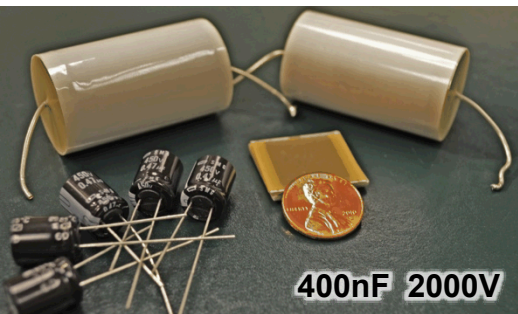
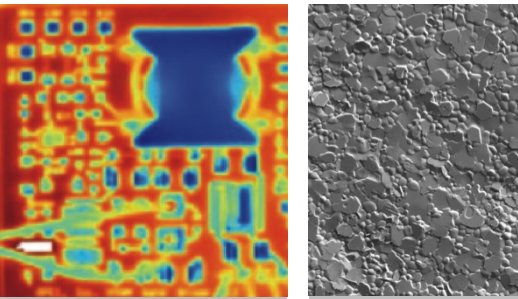


# Ceramic Capacitors for Applications Requiring High Reliability under Challenging Operating Conditions



Geoff Brennecka  
*Sandia National Laboratories*



*Exceptional  
service  
in the  
national  
interest*

The author gratefully acknowledges the support of Dr. Imre Gyuk and the Department of Energy's Office of Electricity Delivery and Energy Reliability.



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# Acknowledgements

## ■ This work would not have been possible without:

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- Prof. Wayne Huebner (MO S&T)
- David Shahin (MO S&T, UMD)
- Ryan Wilkerson (MO S&T, Berkeley)

# Sandia's Work

Shuttle Orbital  
Inspection System



96% of total NW parts



UGS

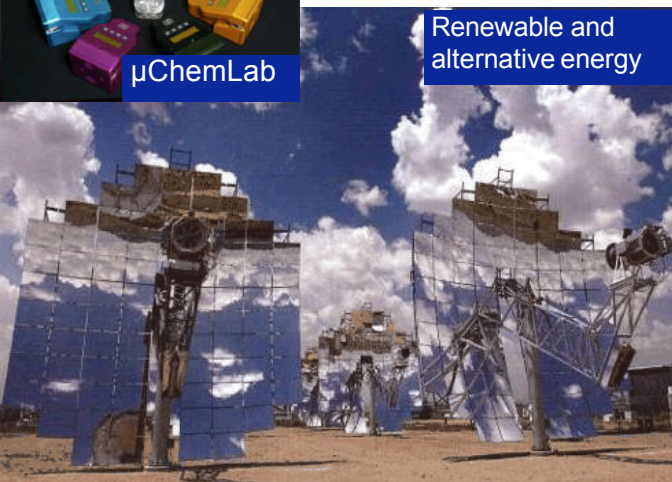
Satellites and  
Surveillance



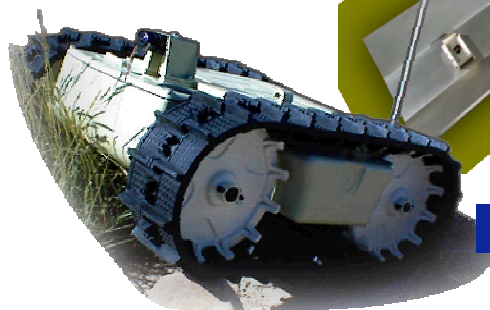
Predator UAV  
with SAR



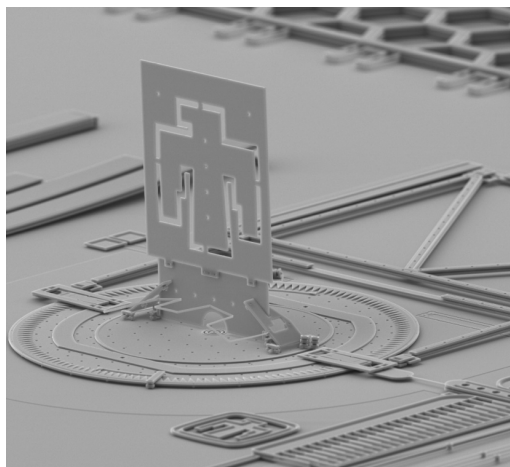
$\mu$ ChemLab



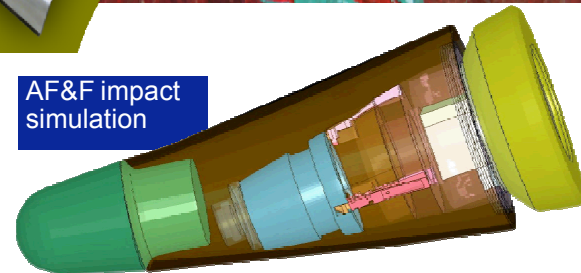
Renewable and  
alternative energy



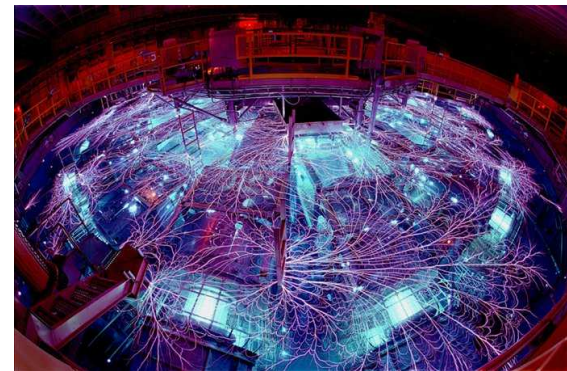
Robotics



Clean room invented at SNL in 1963



AF&F impact  
simulation

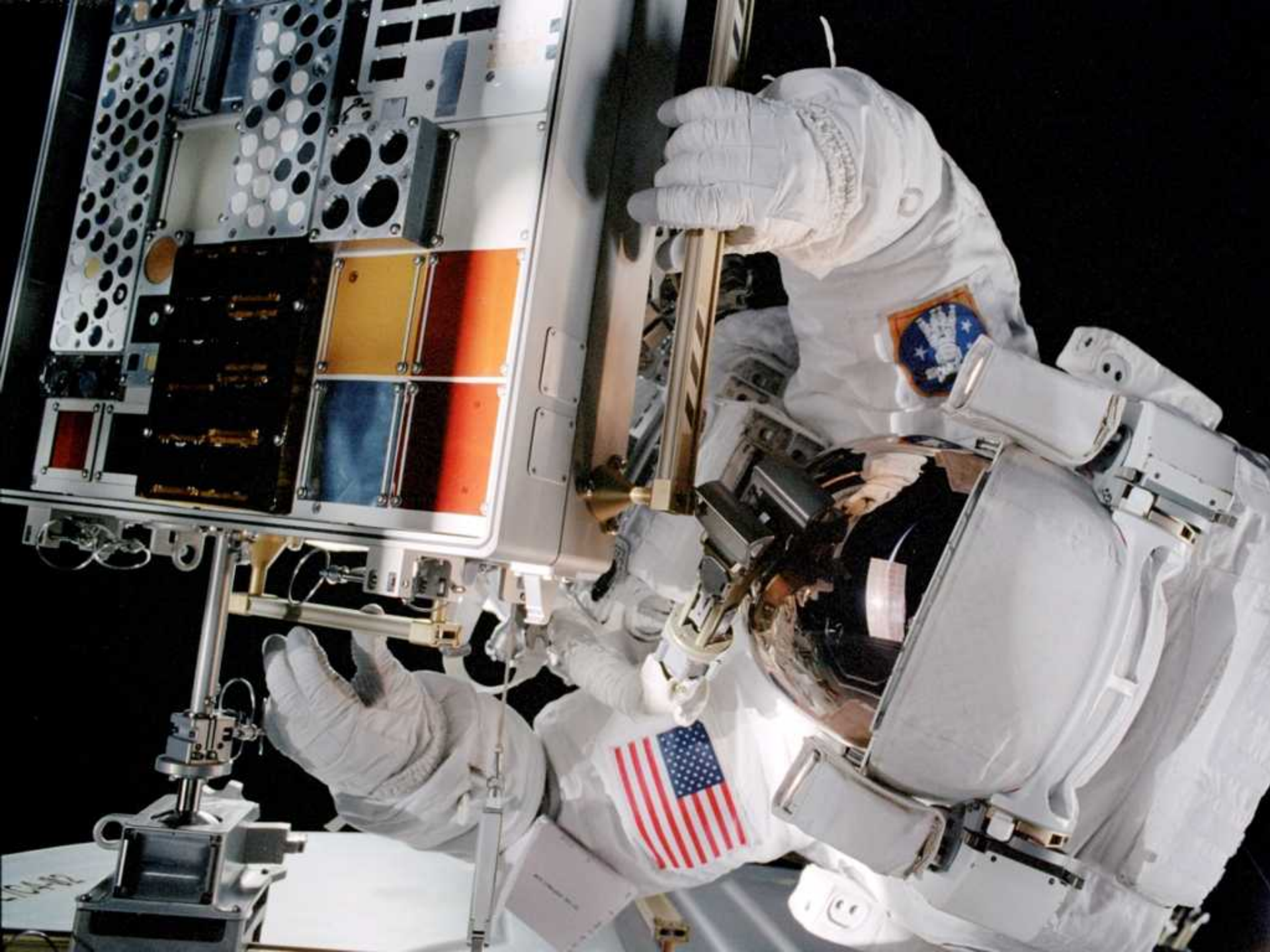


Z machine:  
the world's most powerful X-ray source

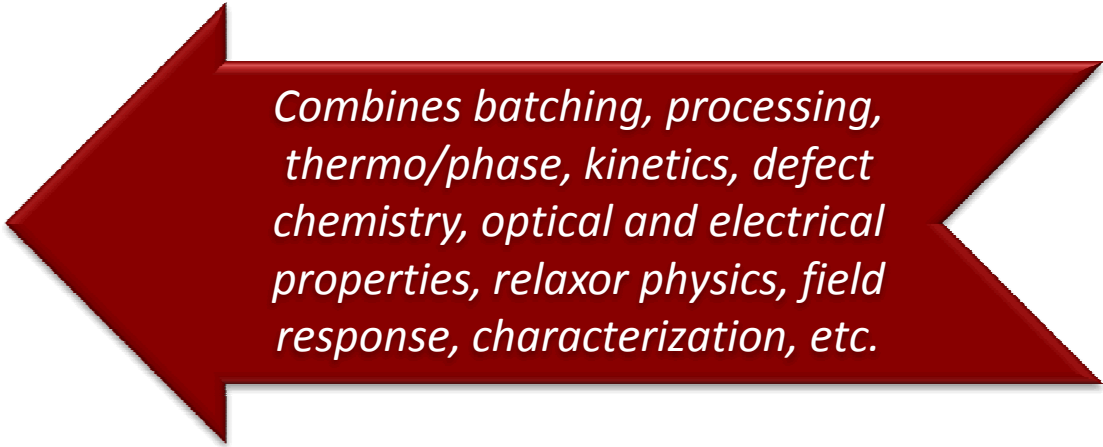








- Refresher on Ferroelectrics and Capacitors
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  - Small volume: high energy density, low cost
  - Long life, reliable operation: high resistivity, high activation energy
- Material Performance
- Demonstration MLCC Fabrication
- Mechanisms
  - Structure
  - Microstructure
  - Processing
  - Defects



*Combines batching, processing, thermo/phase, kinetics, defect chemistry, optical and electrical properties, relaxor physics, field response, characterization, etc.*



# Anatomy of a Typical Paper...

## INTRODUCTION

Capacitors are some sort of magical devices that are key to solving global warming<sup>1</sup>, war<sup>2-3</sup>, famine<sup>4</sup>, and delayed flights<sup>5-6</sup>. Recently, other researchers have suggested that superduper capacitors may even be useful for buzzword<sup>7</sup> and another totally made up phrase<sup>8</sup>. My funding agency is especially fond of them because they have been funding this work for many years with nothing commercial to show for it, so these papers are how I justify continuing the gravy train<sup>9-12</sup>. Smith and Jones have also published in this area; I disagree with their conclusions, but I'm citing their papers because that's the first thing they'll look for if they review this.<sup>13-16</sup> The work from Andrews *et al.* is unrelated, but Tim will buy me a beer at the next conference if I cite him<sup>17</sup>.

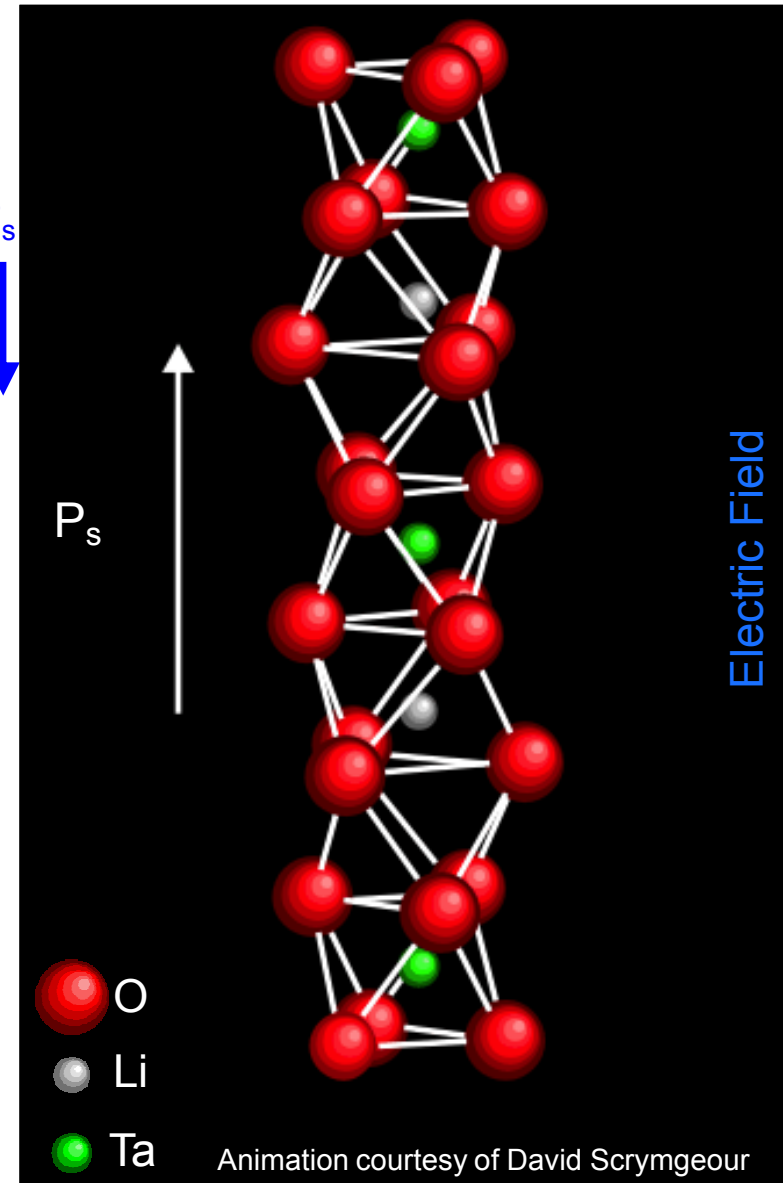
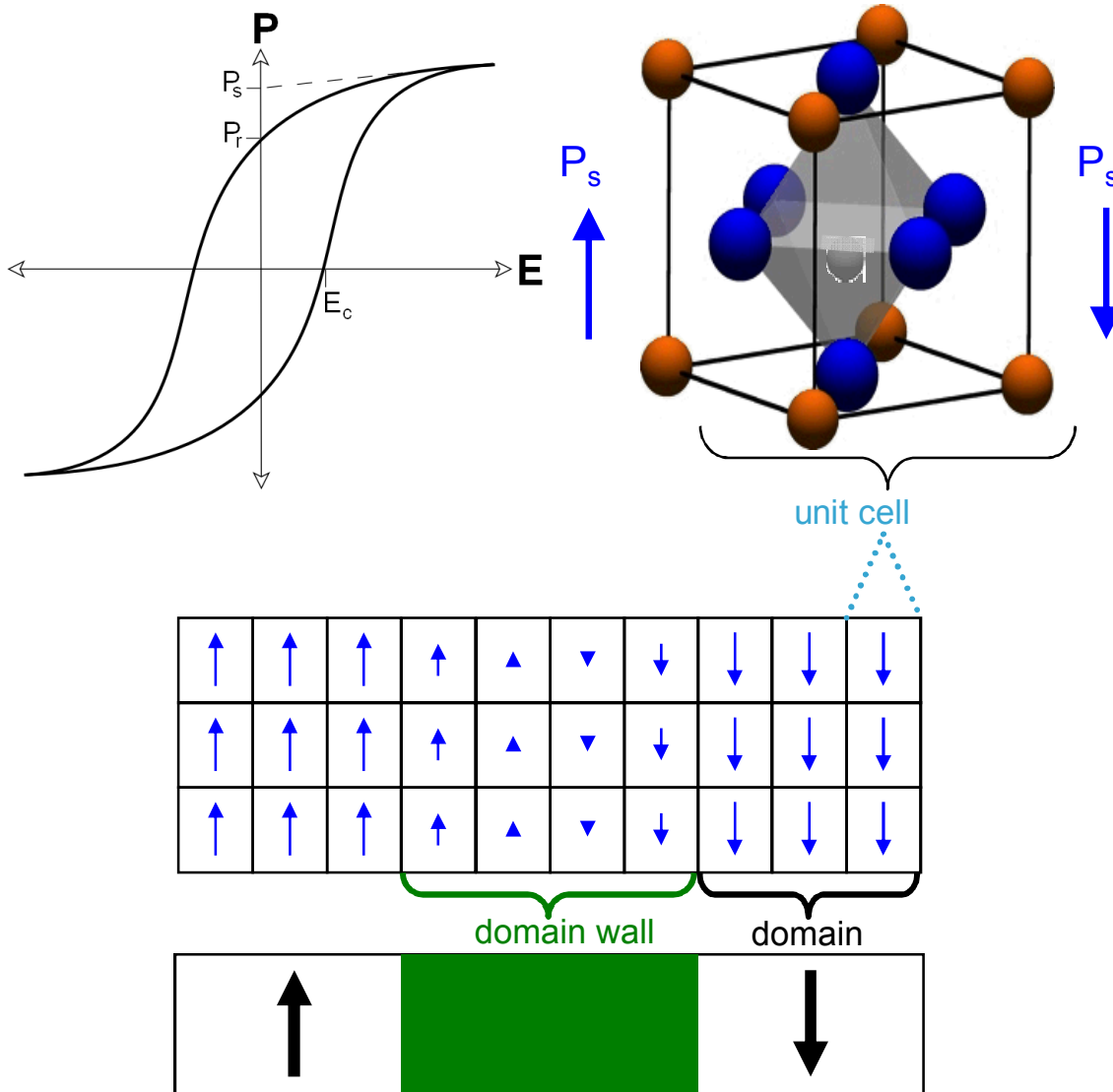
## EXPERIMENTAL

Parts were formed by typical ceramic methods. Processing is boring and doesn't matter anyway. Measurements were made using some home-built equipment that I don't understand and some commercial equipment that I *really* don't understand, so I pretend they're all black boxes that give me data I like.

## RESULTS

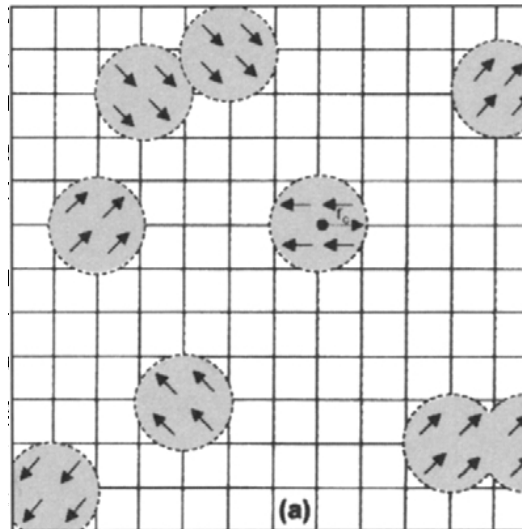
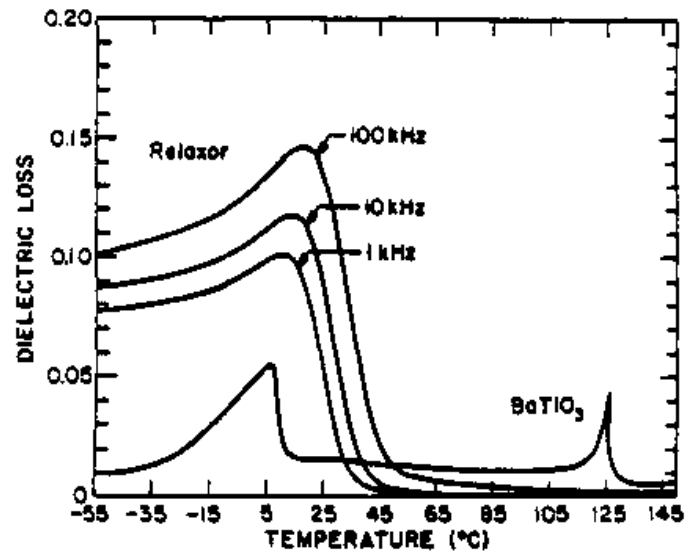
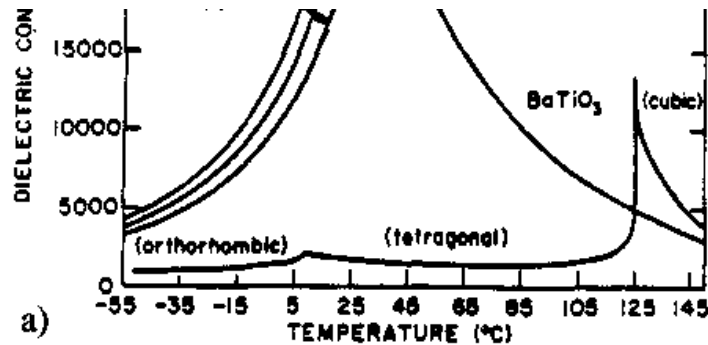
Here's some data. We don't know what it means, but we hope that you're just looking at the figures and title anyway. In fact, it's probably better if you don't pore over every last word of this manuscript. It is readily apparent from the small shoulder on the XRD peak in Fig 1 that we have formed the world's first spontaneous ultralattice. We didn't do any other characterization because it's expensive and slow and never tells us what we want to hear. In Fig 2 we show capacitance data; our instrument gave us other numbers too, but they were weird, so we ignored them...

# Ferroelectrics

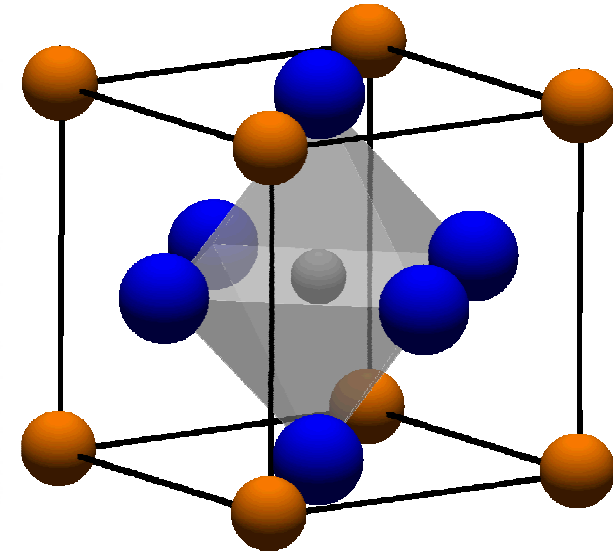




# Relaxor Ferroelectrics



Samara, 2001

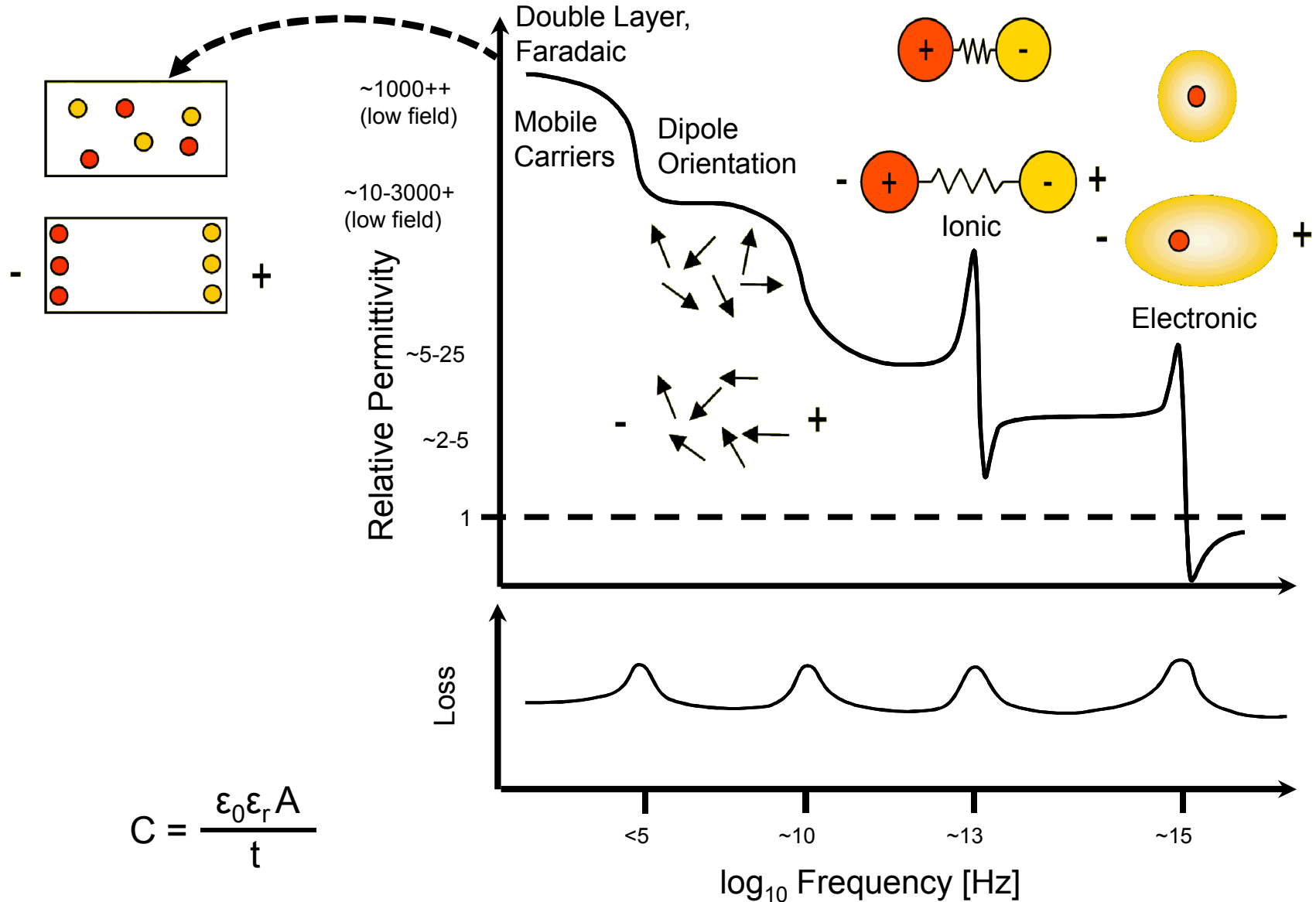


- Prototype is  $\text{Pb}(\text{Mg}_{1/3}\text{Nb}_{2/3})\text{O}_3$
- Disorder (on B-site) and highly polarizable matrix lead to polar nanoregions
- Characteristic frequency dispersion in permittivity and loss maxima
- Dielectric response highly sensitive to pressure, electric field

Table I. Property differences between

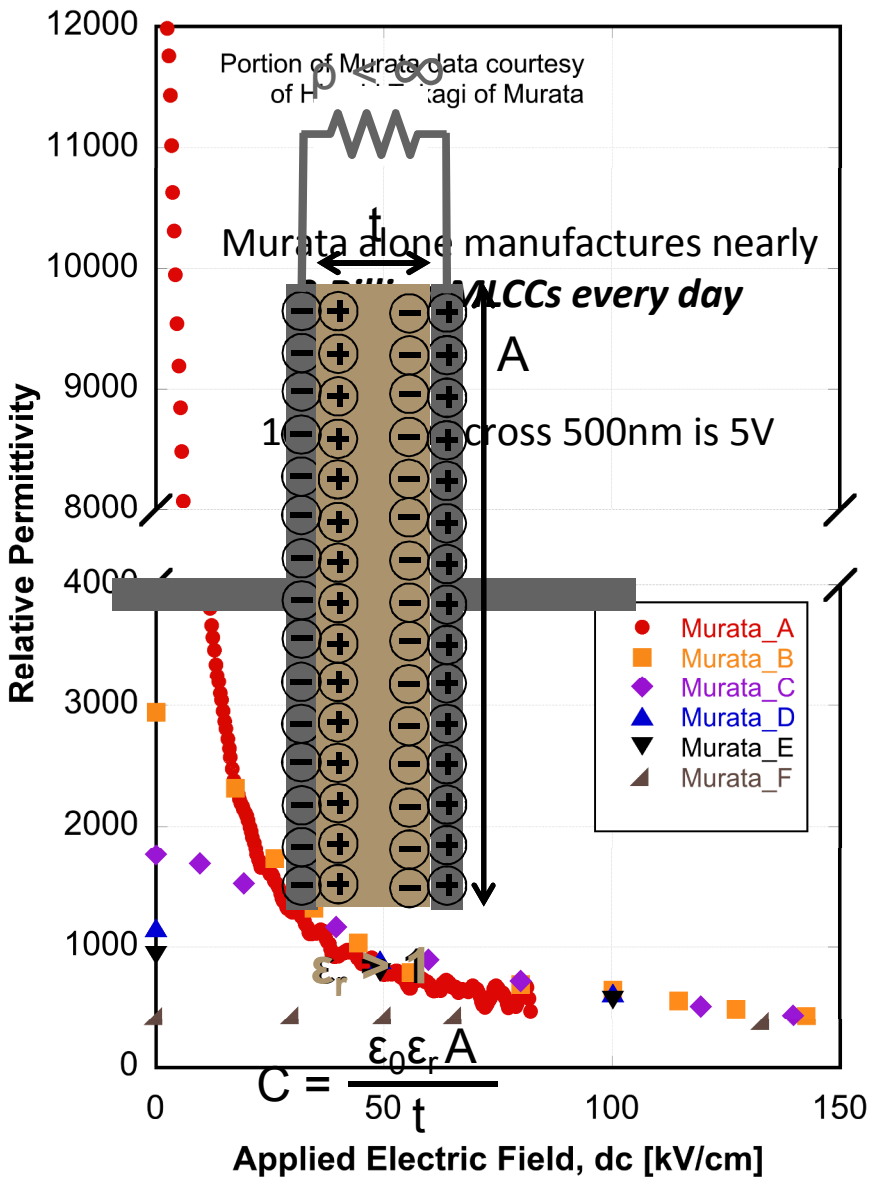
Shrout, ISAF 1990

# Polarization Mechanisms

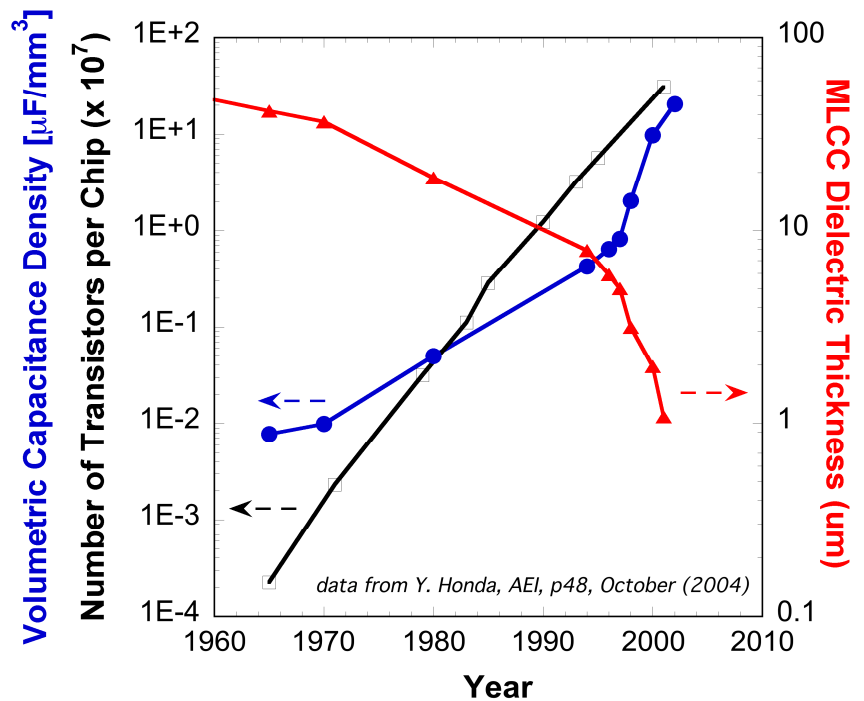
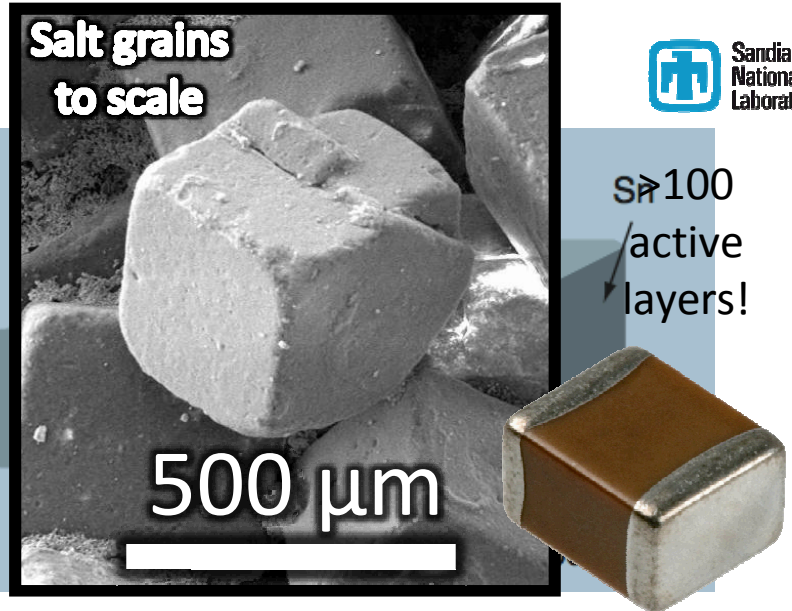




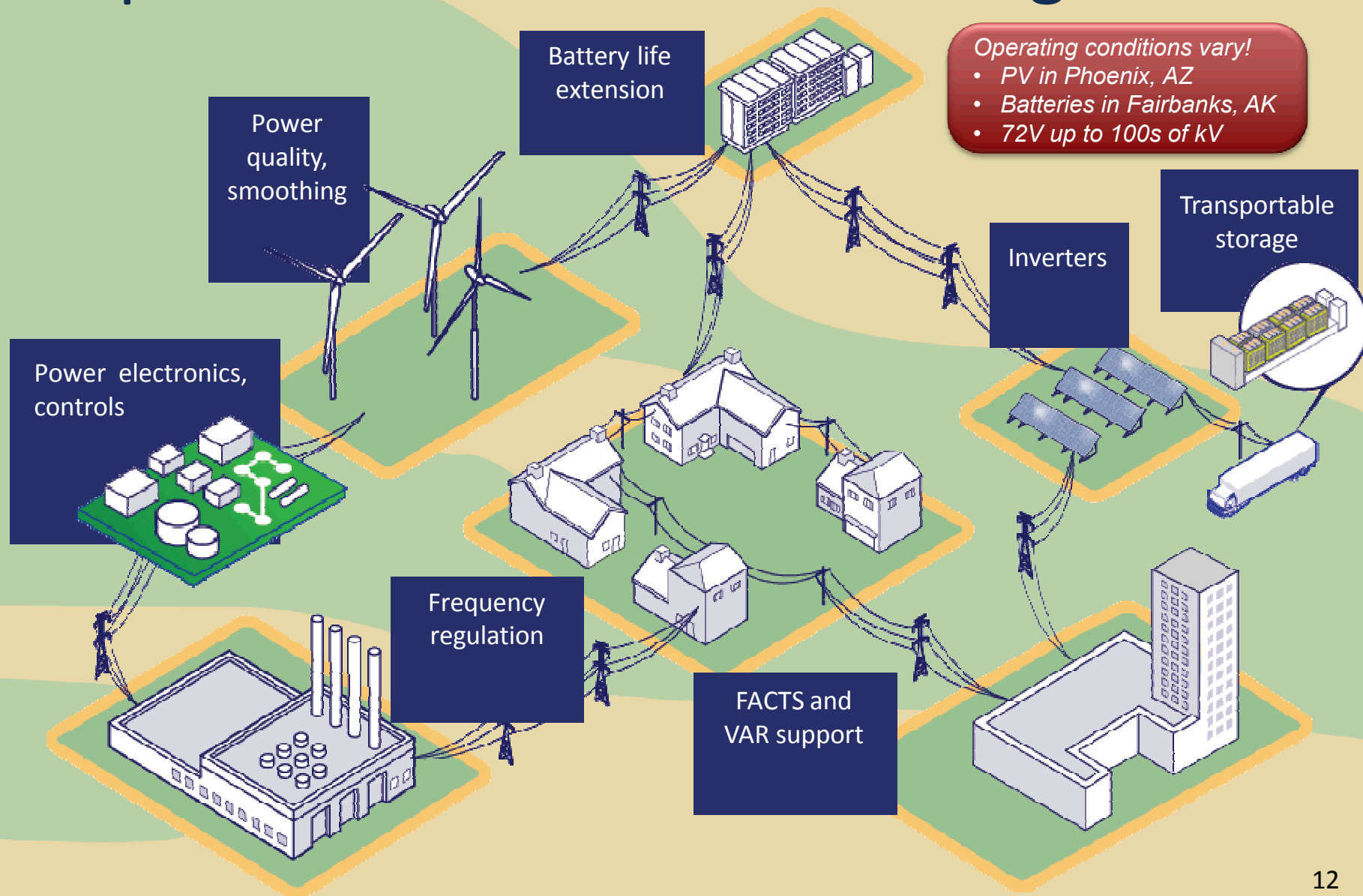
# Capacitors



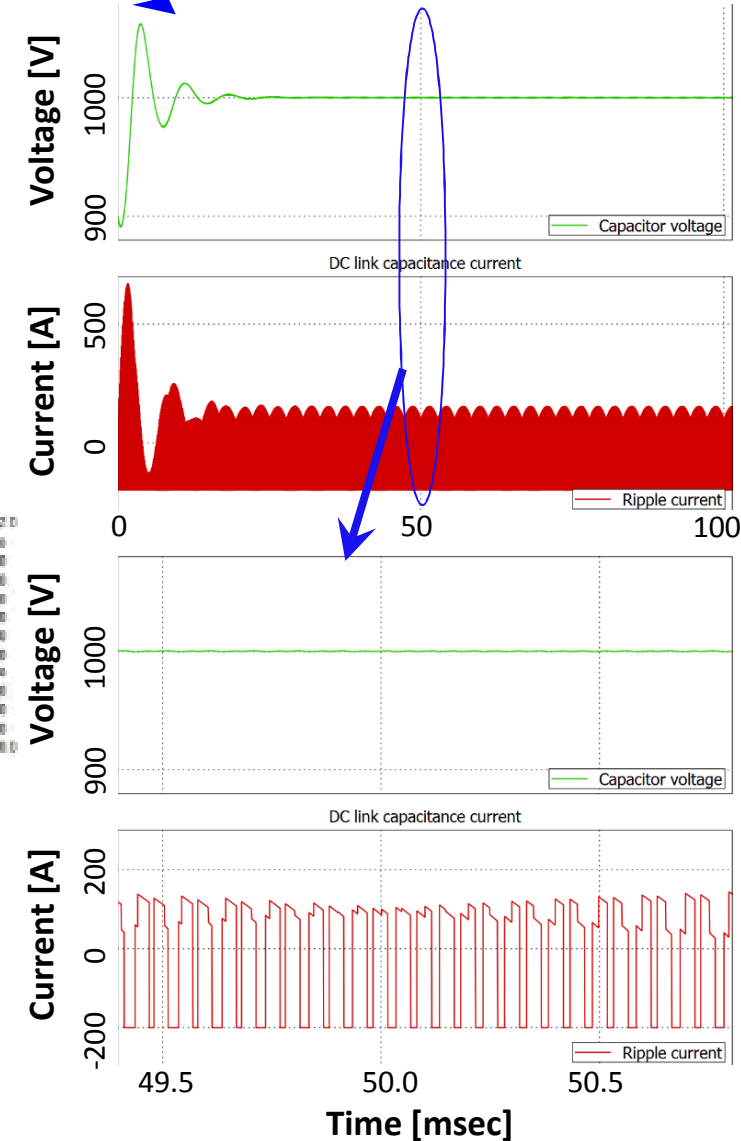
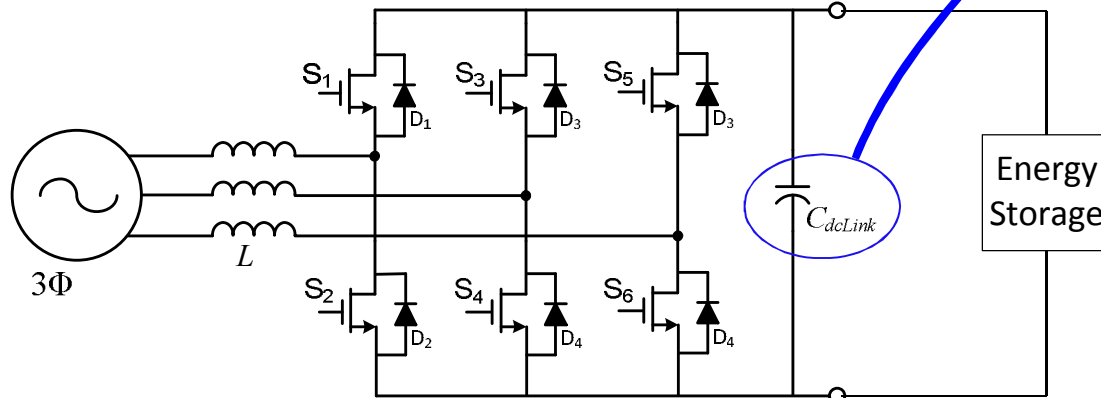
Salt grains  
to scale



# Capacitor Needs for Grid Storage



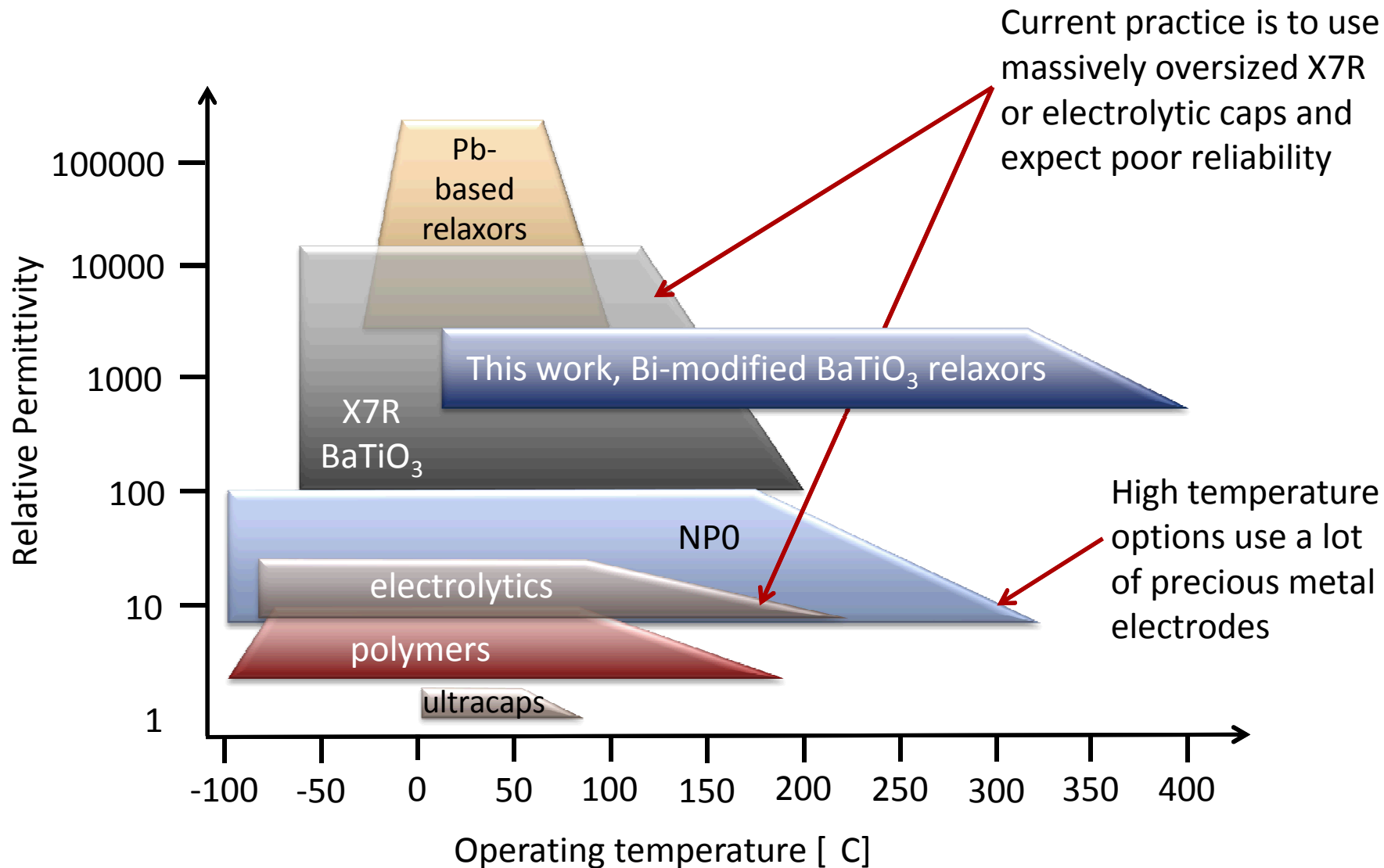
# Application Space



- High energy density, high capacitance density
- Operation  $>200^{\circ}\text{C}$  needed,  $>300^{\circ}\text{C}$  desired
- Low ESR for ripple current and power handling
- Decades of reliable operation



# Available Options

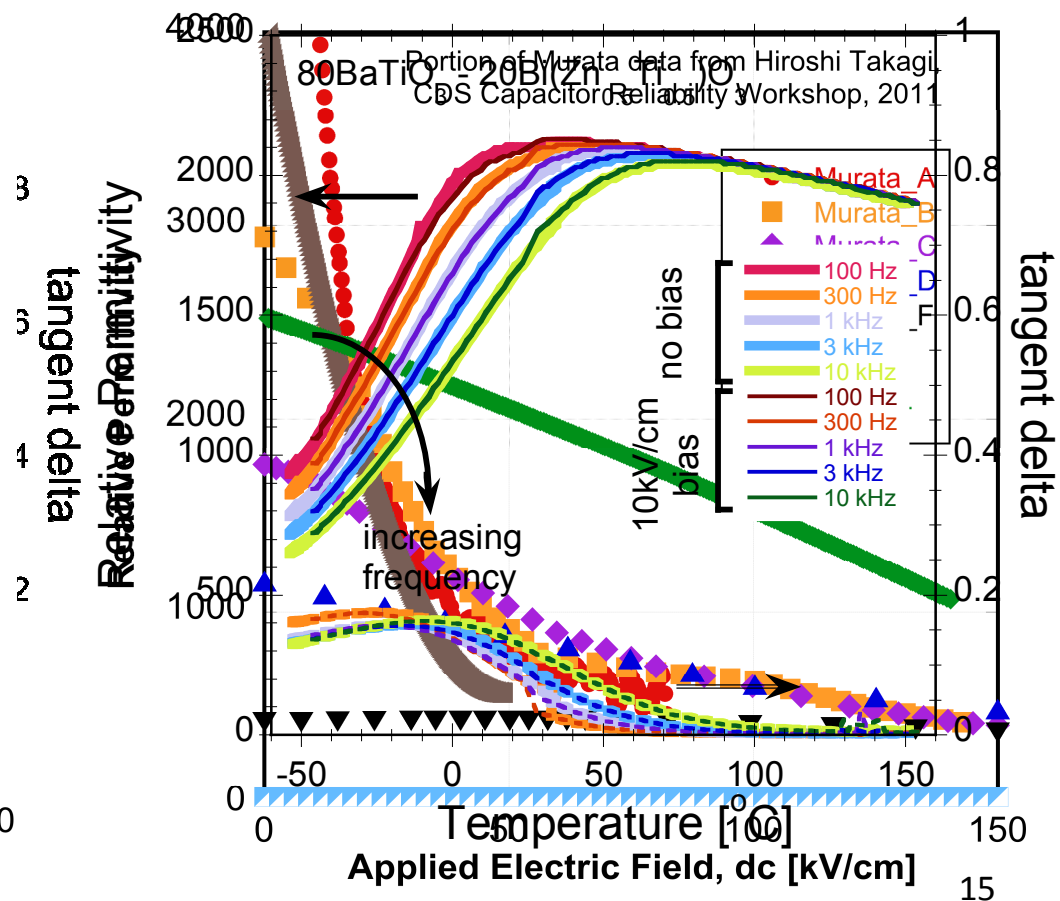
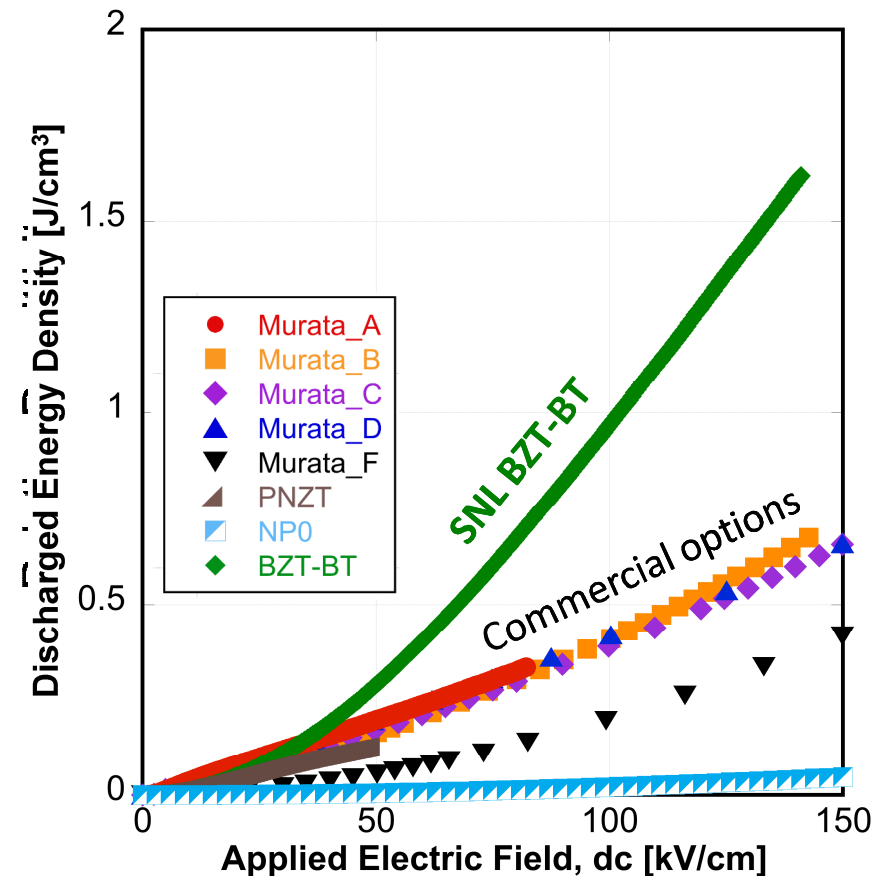


# High Energy Density Dielectrics

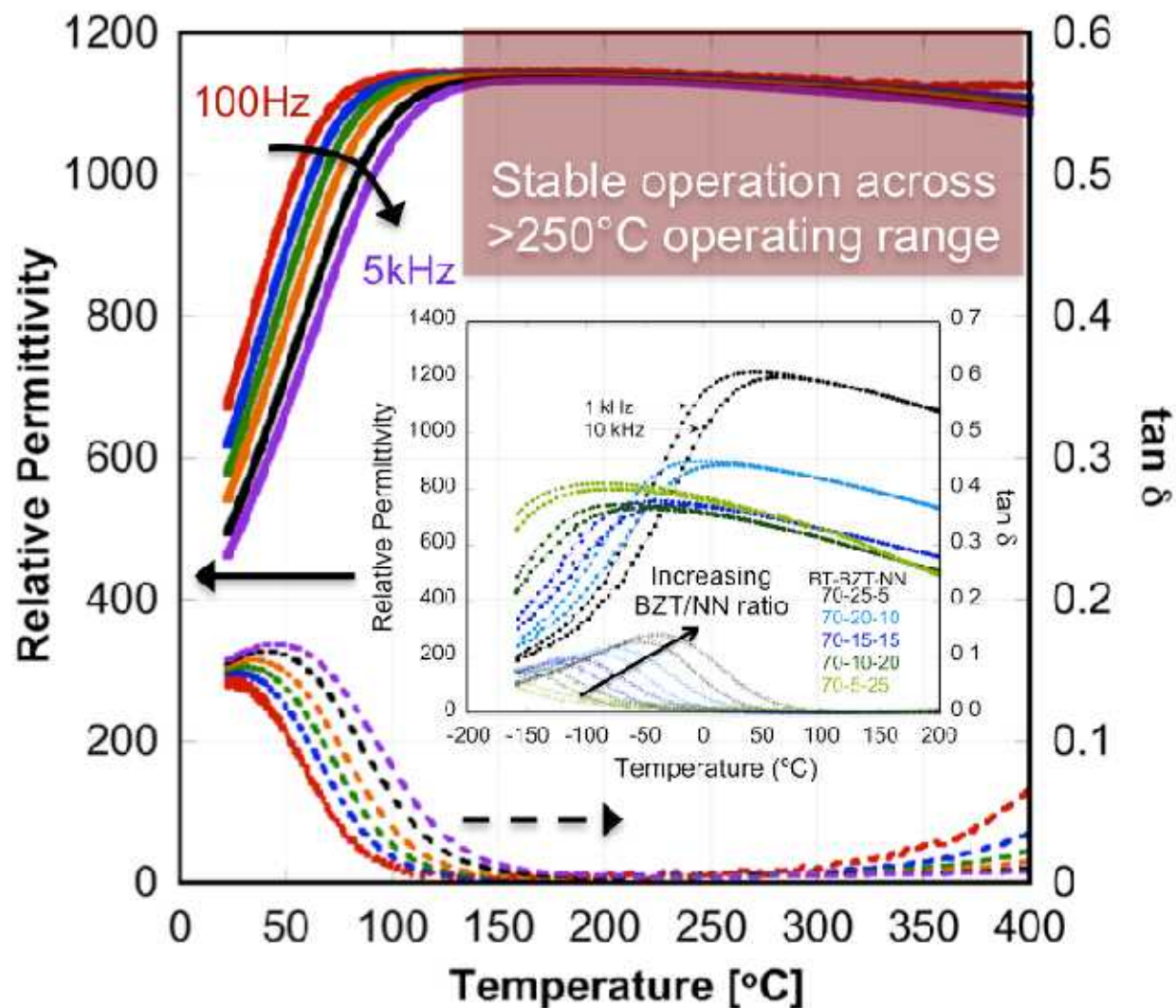
$$J = \int_0^{V_{\max}} CV^2 \rightarrow \int_0^{E_{\max}} KE^2$$

For high-K materials,  $K=K(E)$

→ maintaining high K at high E is important



# High Temperature Operation

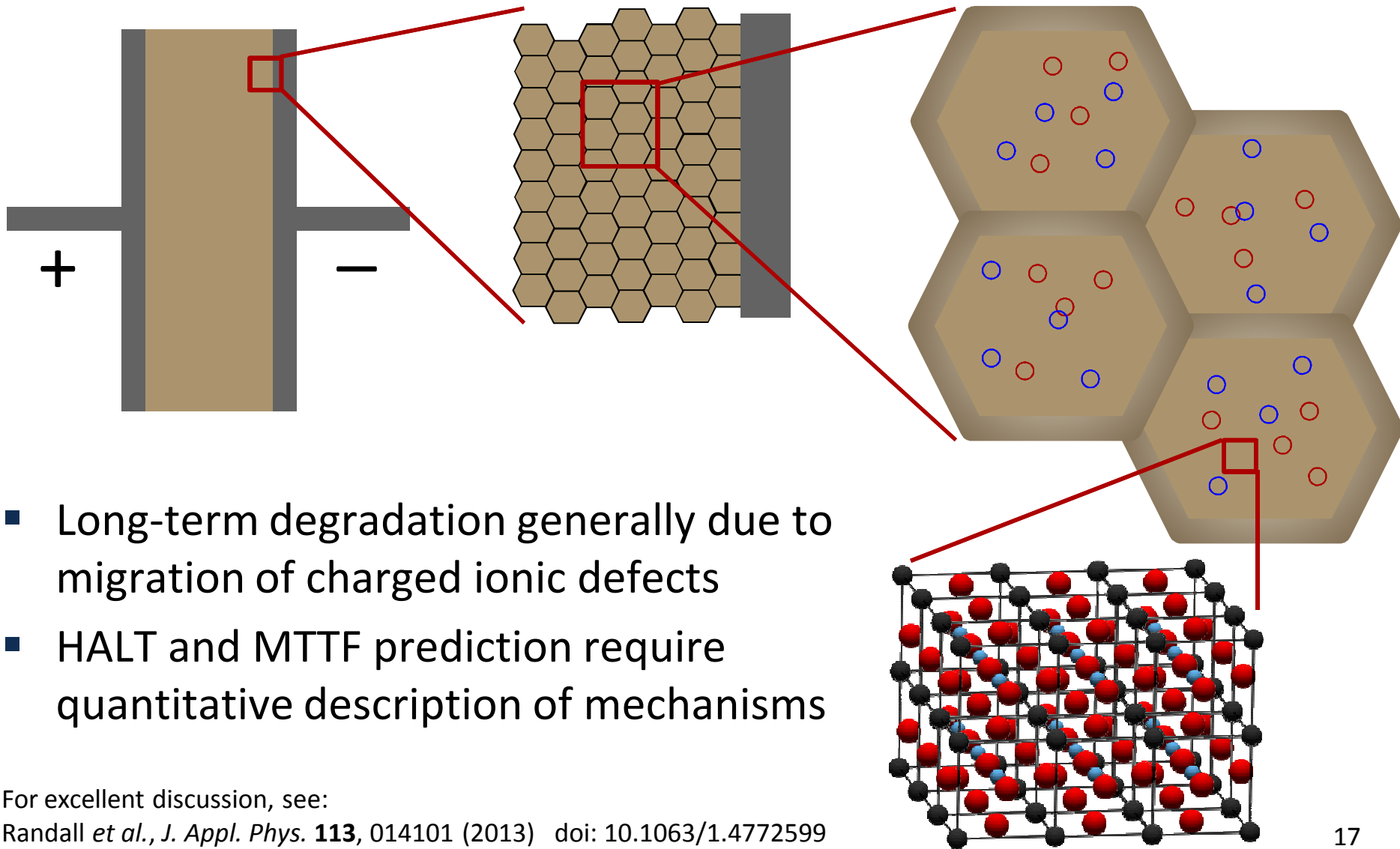


- BiScO<sub>3</sub> stabilizes high temperature permittivity
- SrTiO<sub>3</sub> or NaNbO<sub>3</sub> additions shift relaxor transition to lower temperatures

**We can shift the temperature range for stable operation around by ~250°C via chemical modification *without sacrificing voltage stability***



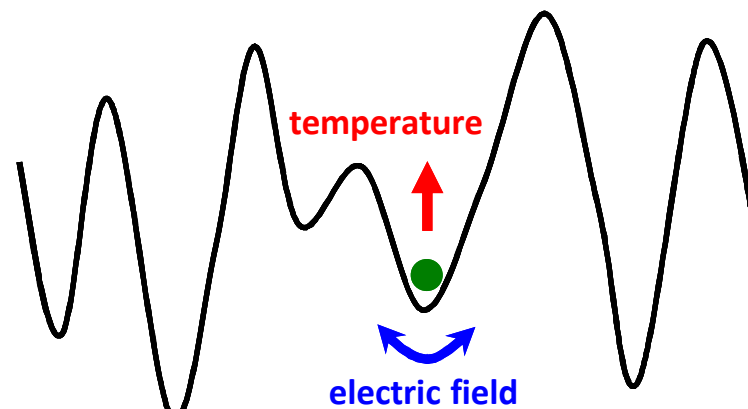
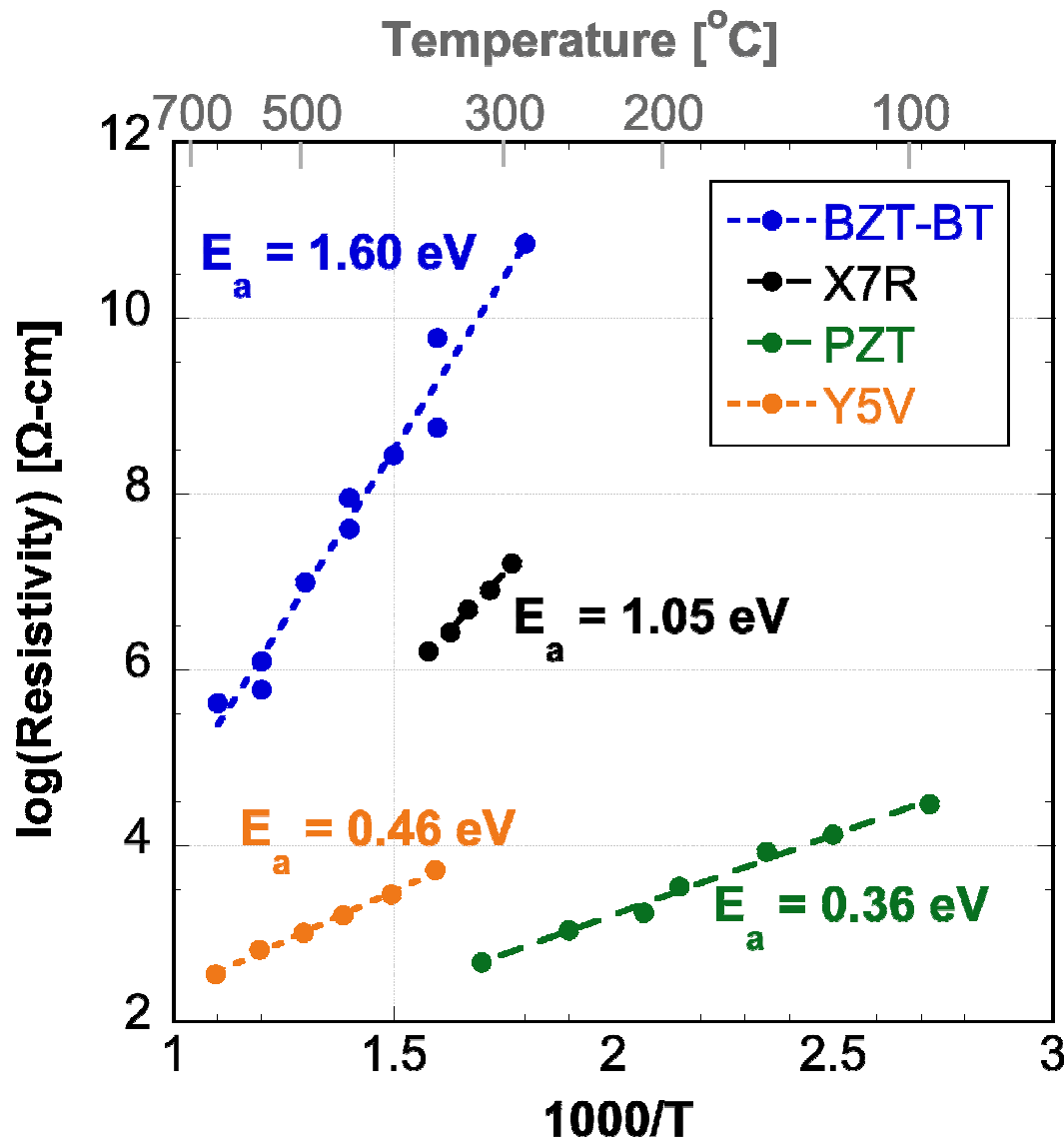
# Degradation in Ceramic Dielectrics



For excellent discussion, see:

Randall *et al.*, *J. Appl. Phys.* **113**, 014101 (2013) doi: 10.1063/1.4772599

# High Resistivity → Reliable



Higher resistivity and larger activation energy for conduction both translate into longer lifetimes and higher reliability, particularly at elevated temperatures.

# Outline

- Refresher on Ferroelectrics and Capacitors
- Application drivers:
  - High operating temperatures, high power: low ESR, low loss
  - Small volume: high energy density, low cost
  - Long life, reliable operation: high resistivity, high activation energy
- Material Performance
- **Demonstration Fabrication**
- Mechanisms
  - Structure
  - Microstructure
  - Processing
  - Defects



# Prototype MLCCs: 200nF @ 1700V

BaCO<sub>3</sub>

ZnO

Bi<sub>2</sub>O<sub>3</sub>

TiO<sub>2</sub>

Mix/mill, calcine @ 950°C, Mill

*Single phase by XRD*

# Prototype MLCCs: 200nF @ 1700V

$\text{BaCO}_3$

$\text{ZnO}$

$\text{Bi}_2\text{O}_3$

$\text{TiO}_2$

Mix/mill, calcine @ 950°C, Mill

*Single phase by XRD*

Bind, Press,  
Sinter @  
1120-1180°C

The image cannot be displayed. Your computer may not have enough memory to open the image, or the image may have been corrupted. Restart your computer, and then open the file again. If the red x still appears, you may have to delete the image and then insert it again.

# Prototype MLCCs: 200nF @ 1700V

$\text{BaCO}_3$

$\text{ZnO}$

$\text{Bi}_2\text{O}_3$

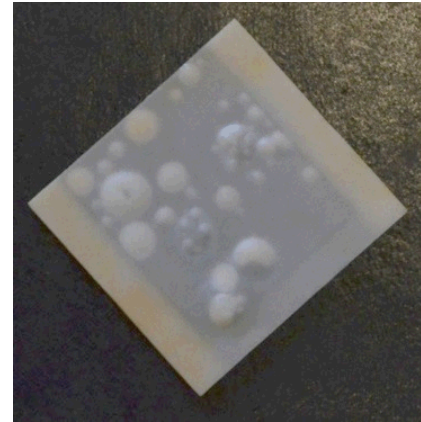
$\text{TiO}_2$

Mix/mill, calcine @  $950^\circ\text{C}$ , Mill

*Single phase by XRD*

Bind, Press,  
Sinter @  
 $1120\text{--}1180^\circ\text{C}$

Cast and co-  
fire with Pt  
@  $1120^\circ\text{C}$

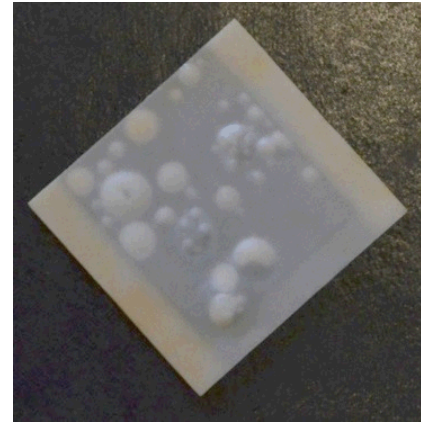
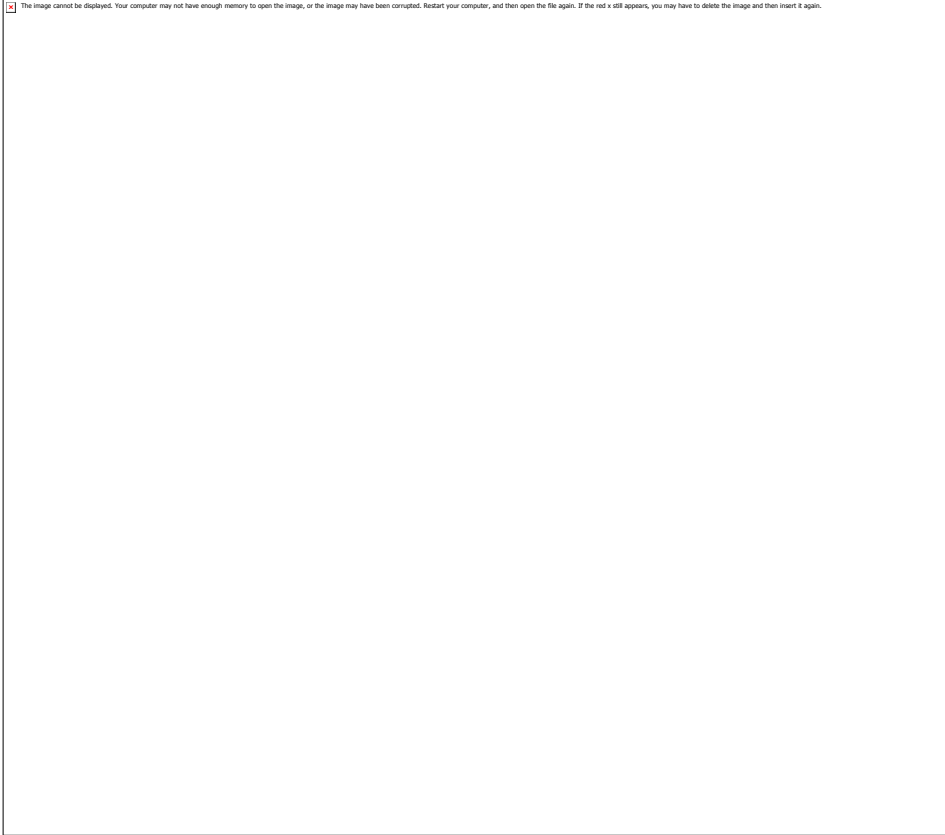


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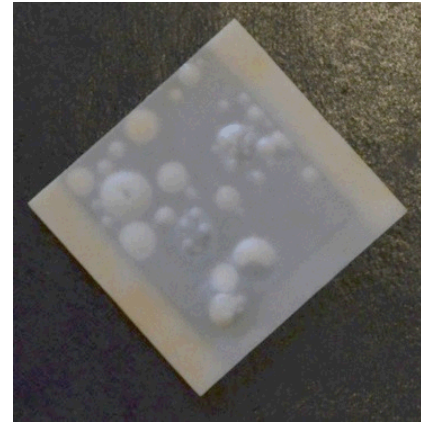
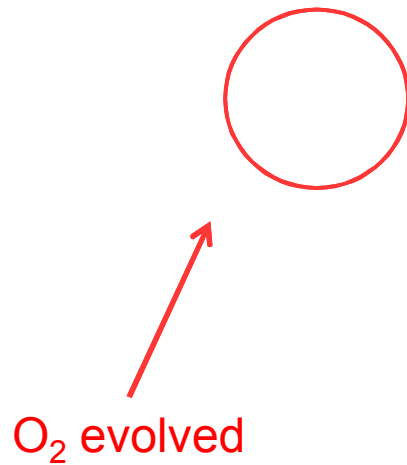
# Prototype MLCCs: 200nF @ 1700V

## MLCC Binder Burnout

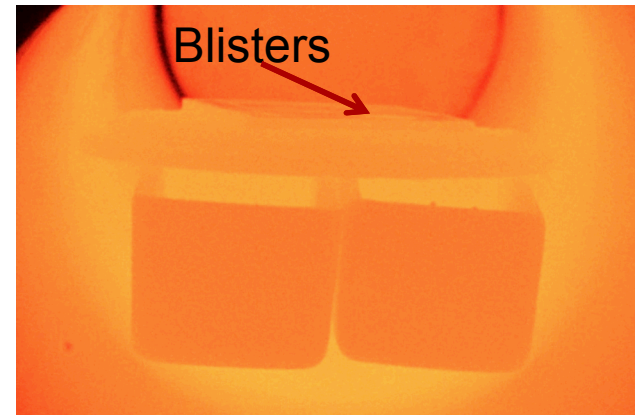


# Prototype MLCCs: 200nF @ 1700V

## MLCC Sintering

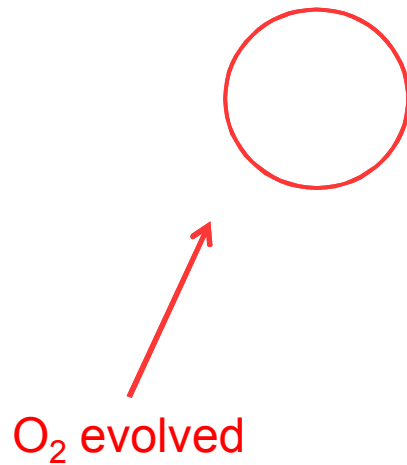


## MLCC sintering at 1210°C



# Prototype MLCCs: 200nF @ 1700V

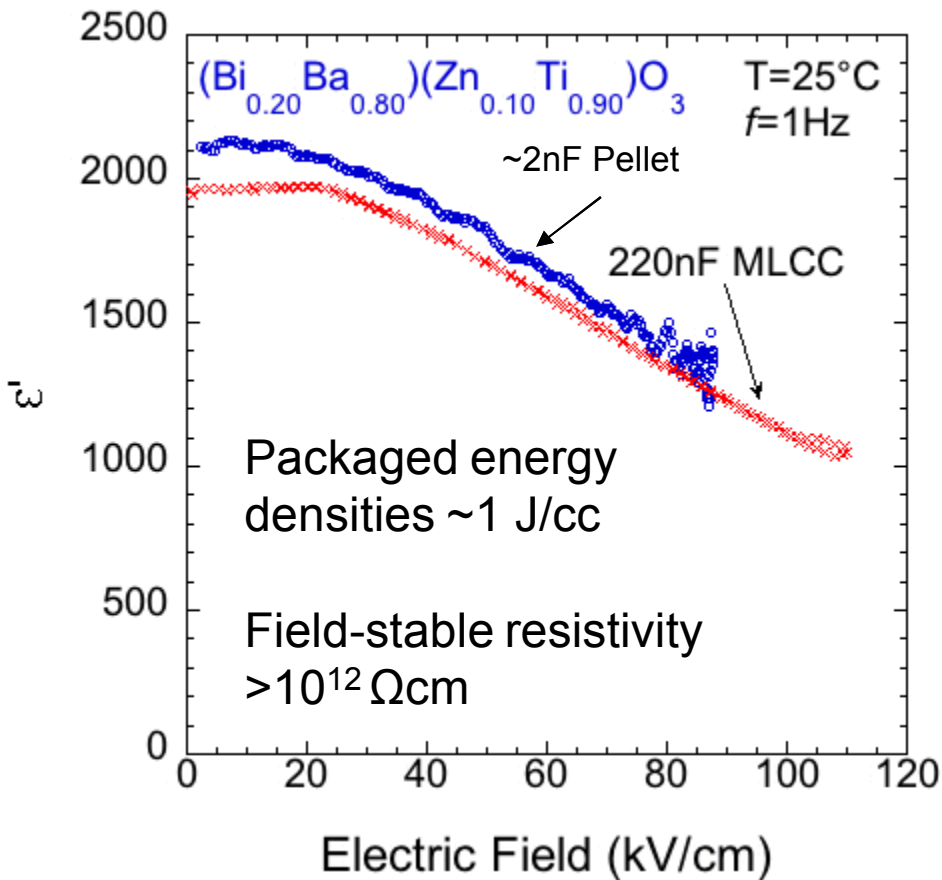
## MLCC Sintering



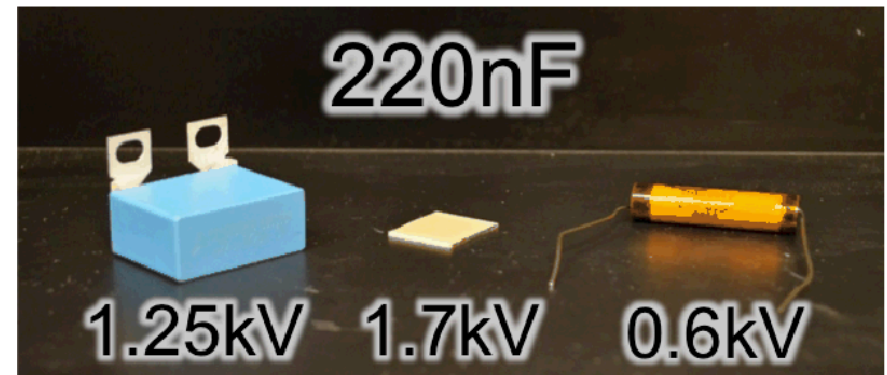
Cast and co-  
fire with Pt  
@ 1040°C



# Prototype MLCCs: 200nF @ 1700V

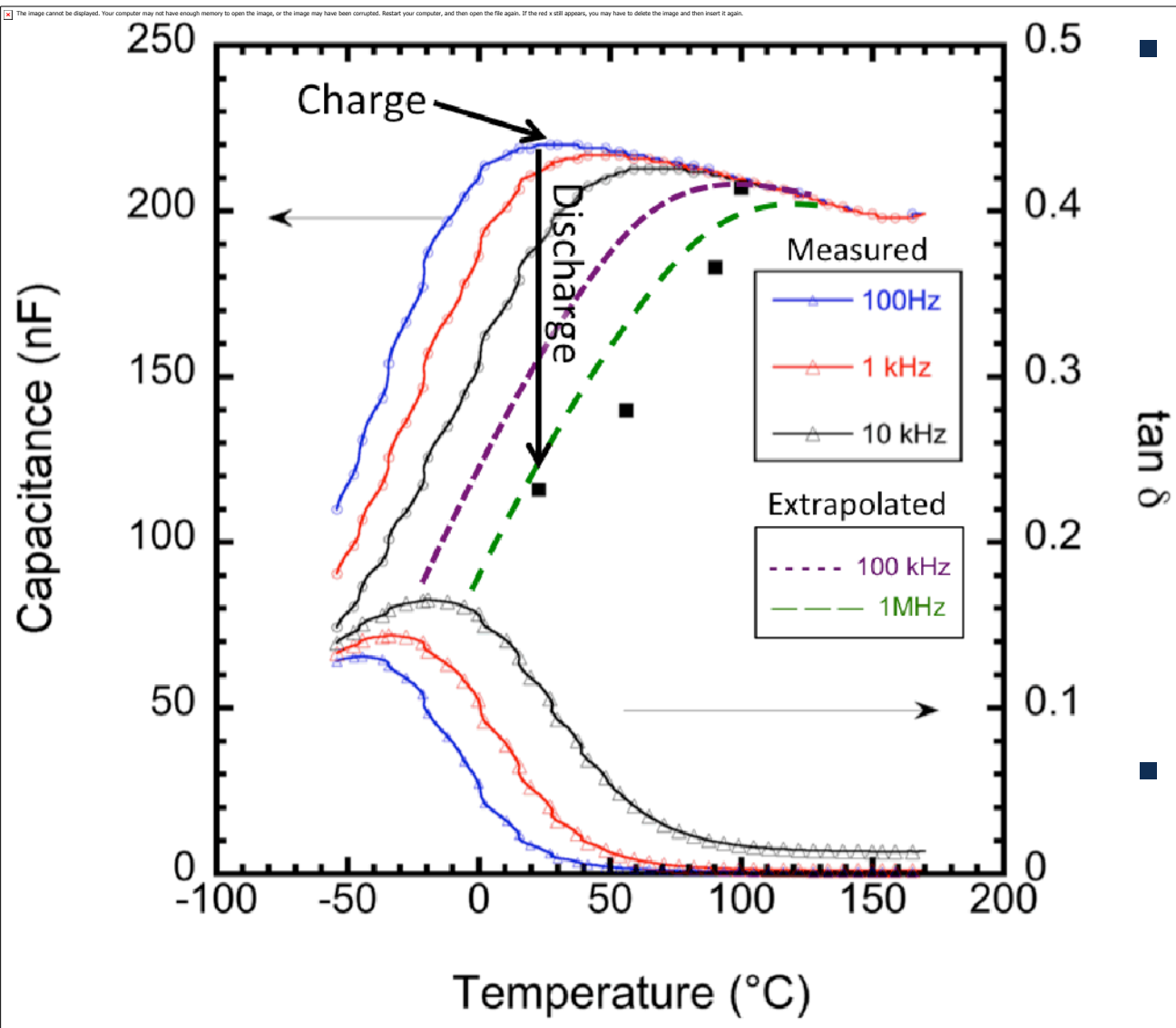


Cast and co-fire with Pt @  $1040^\circ\text{C}$



- Large MLCCs retain performance across operating temperatures, electric fields.
- Mechanism(s)??

# Time Domain Performance



- Relaxor dielectrics exhibit characteristic frequency dispersion over relatively broad temperature ranges
  - For switched inverter designs which charge slowly and discharge quickly at irregular intervals, which values are relevant?
- Direct time-domain measurements map well to frequency domain data

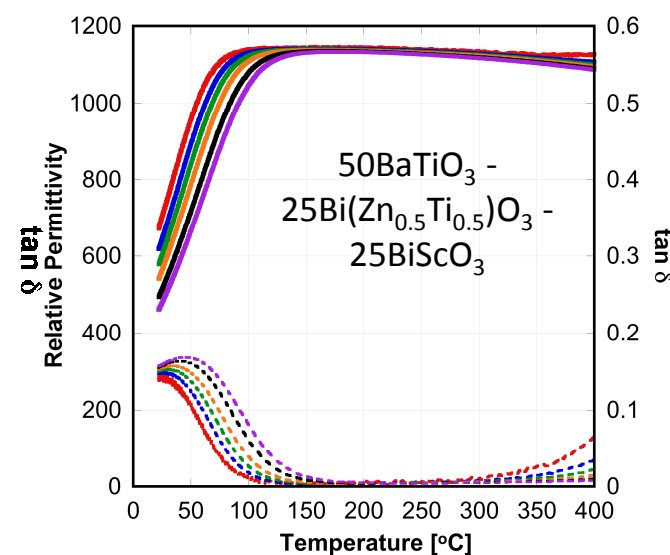
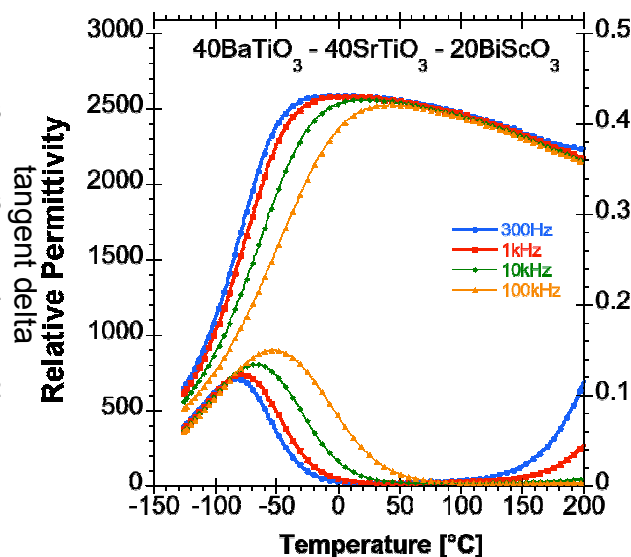
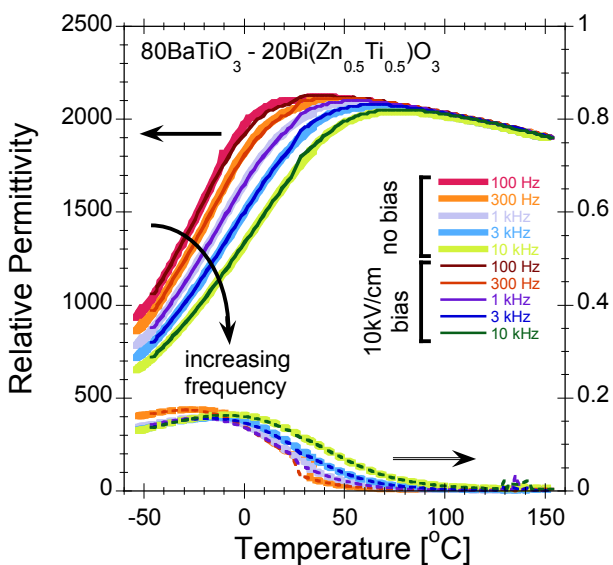
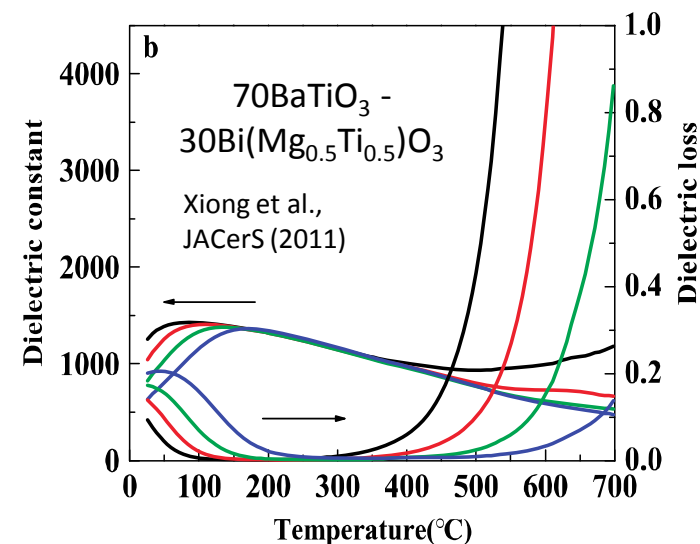
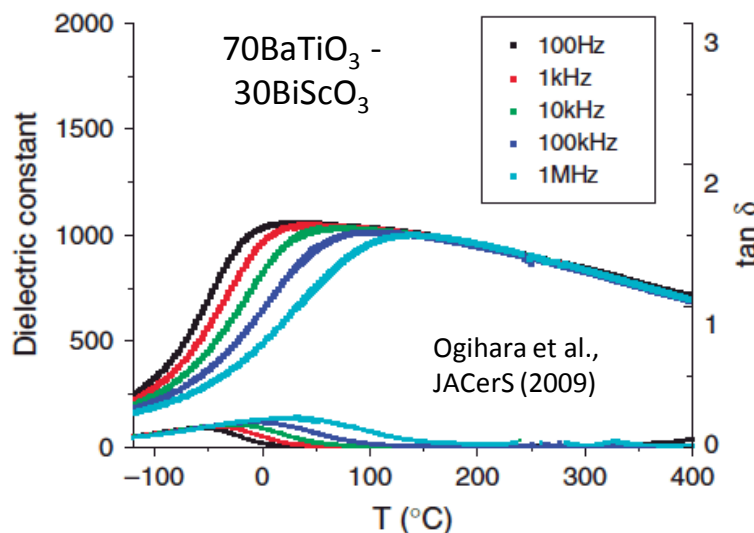


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  - Microstructure
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  - Defects

# Bi-modified BaTiO<sub>3</sub> Relaxors

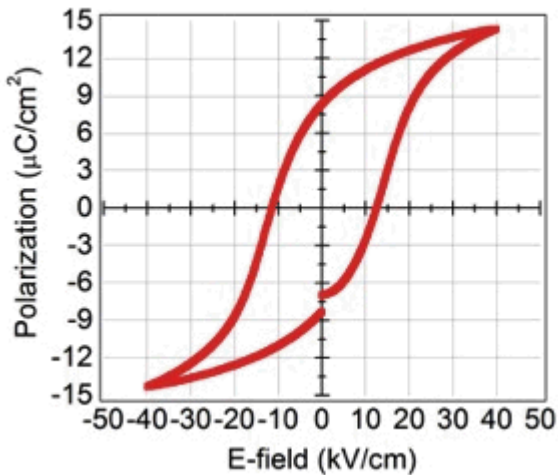
BaTiO<sub>3</sub> +

- Bi(Zn<sub>0.5</sub>Ti<sub>0.5</sub>)O<sub>3</sub>
- Bi(Mg<sub>0.5</sub>Ti<sub>0.5</sub>)O<sub>3</sub>
- BiScO<sub>3</sub>
- BiFeO<sub>3</sub>
- BiInO<sub>3</sub>
- Bi(Ni<sub>0.5</sub>Ti<sub>0.5</sub>)O<sub>3</sub>
- ...



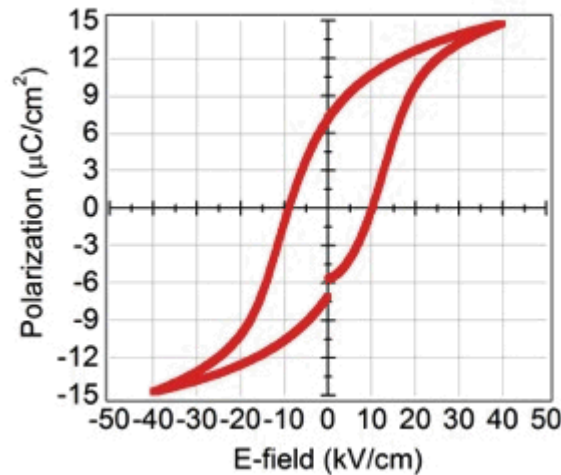
# $x\text{BaTiO}_3 - (1-x)\text{Bi}(\text{Zn}_{0.5}\text{Ti}_{0.5})\text{O}_3$

P-E hysteresis of 0.95BT-0.05BZT



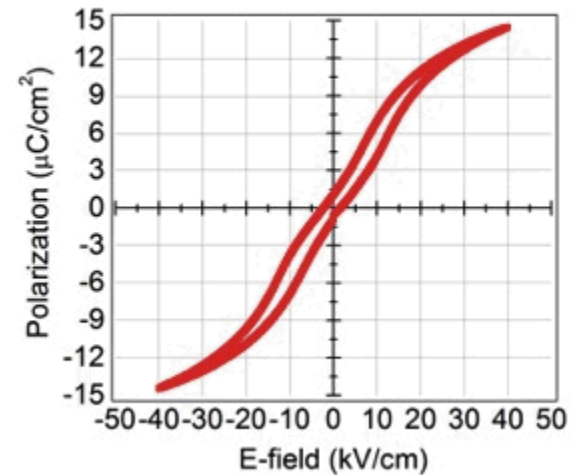
$x = 0.95$

P-E hysteresis of 0.93BT-0.07BZT



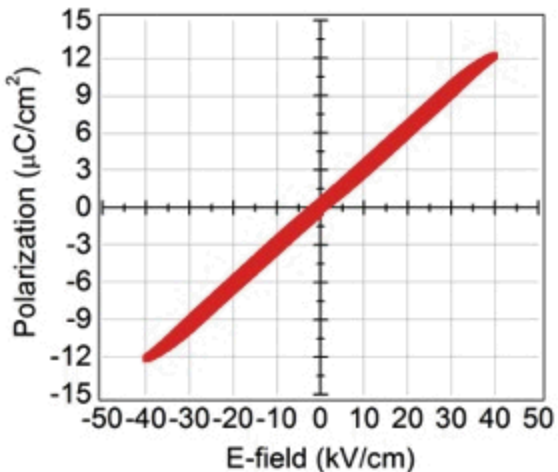
$x = 0.93$

P-E hysteresis of 0.92BT-0.08BZT



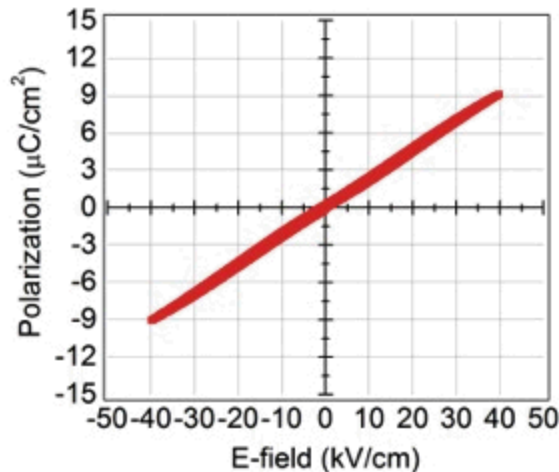
$x = 0.92$

P-E hysteresis of 0.91BT-0.09BZT



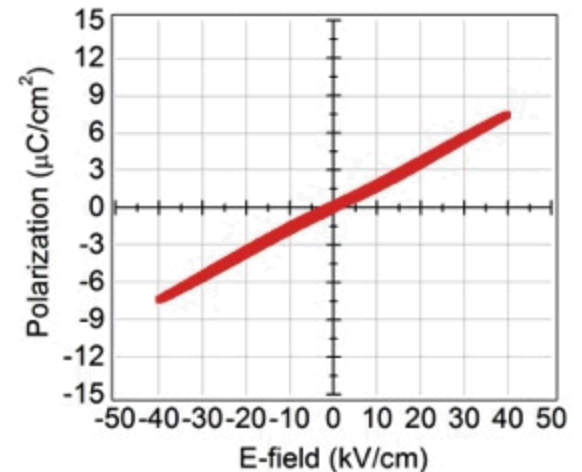
$x = 0.91$

P-E hysteresis of 0.89BT-0.11BZT



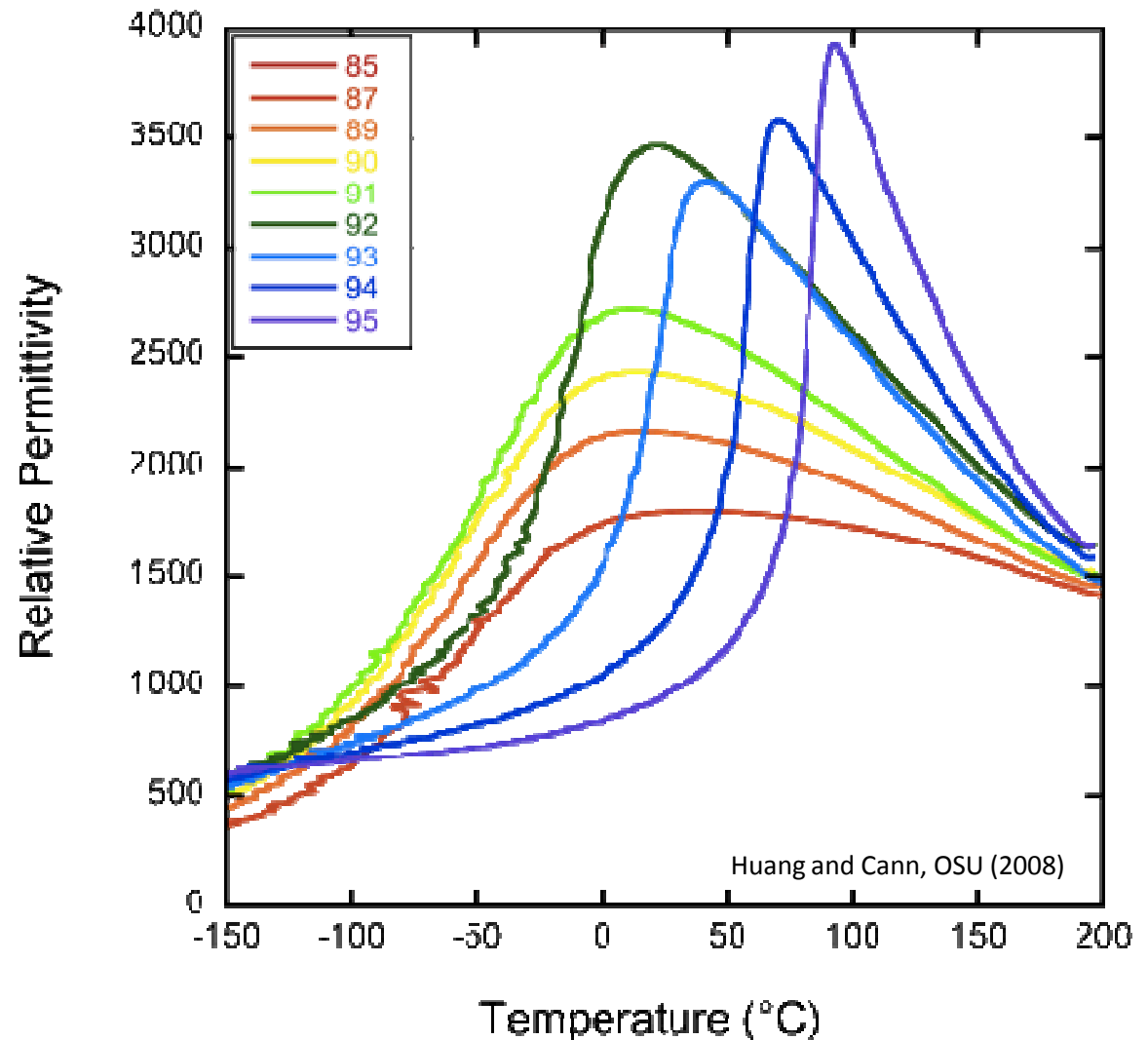
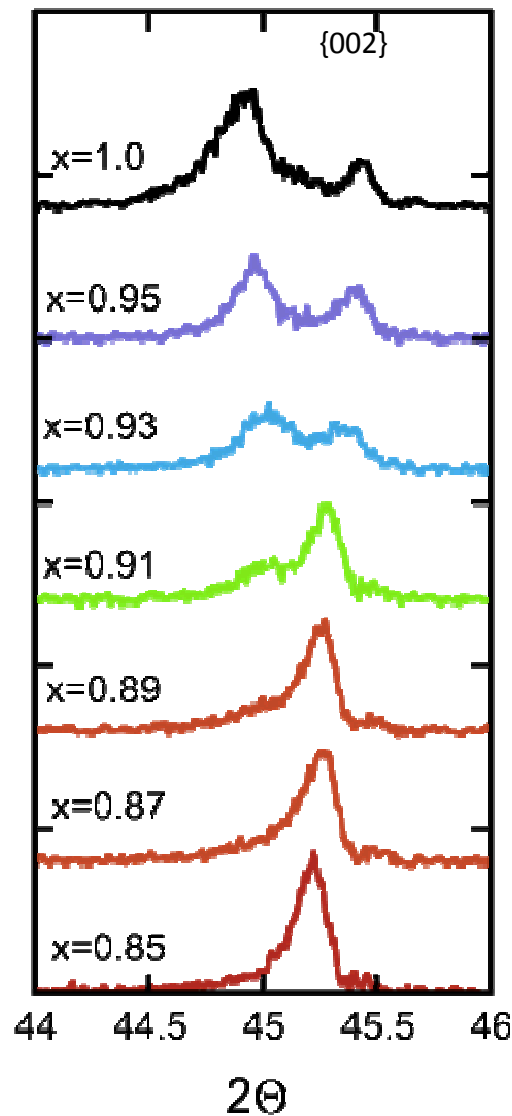
$x = 0.89$

P-E hysteresis of 0.87BT-0.13BZT

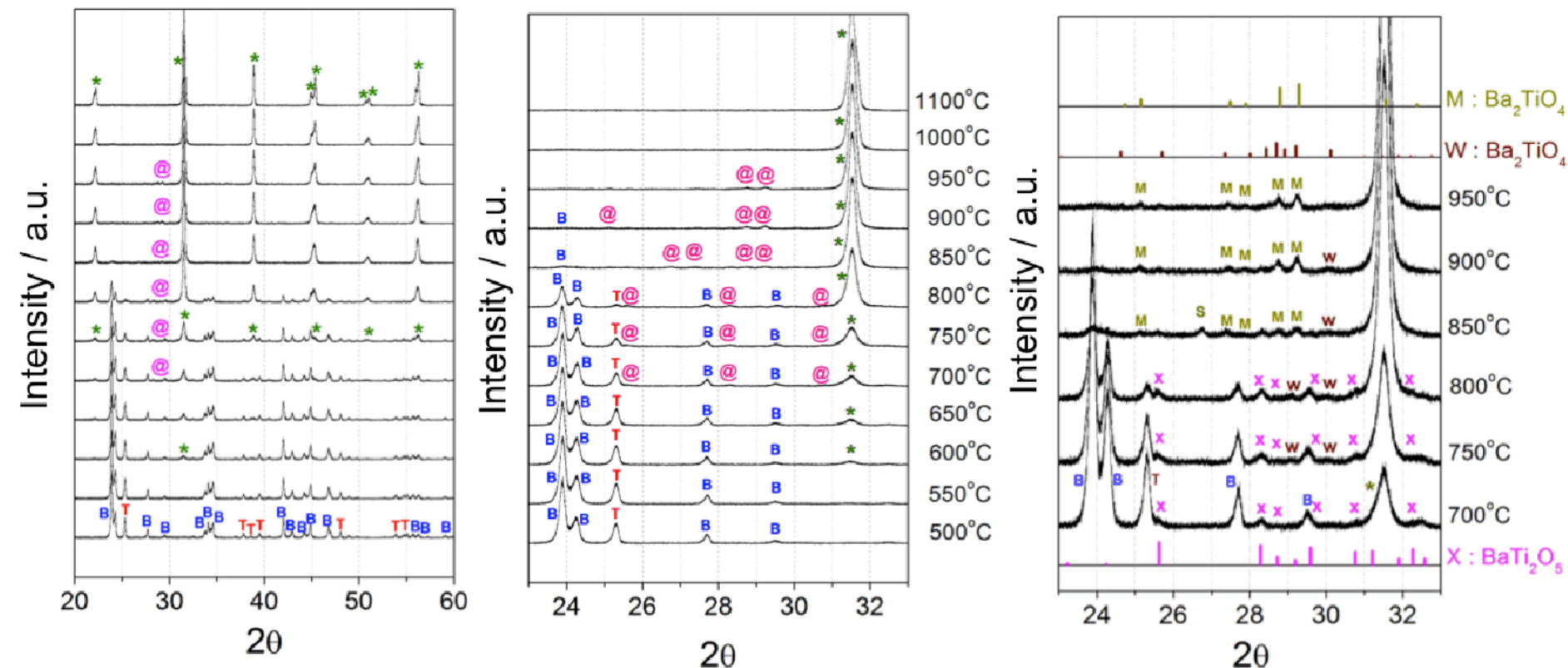


$x = 0.87$

# $x\text{BaTiO}_3 - (1-x)\text{Bi}(\text{Zn}_{0.5}\text{Ti}_{0.5})\text{O}_3$



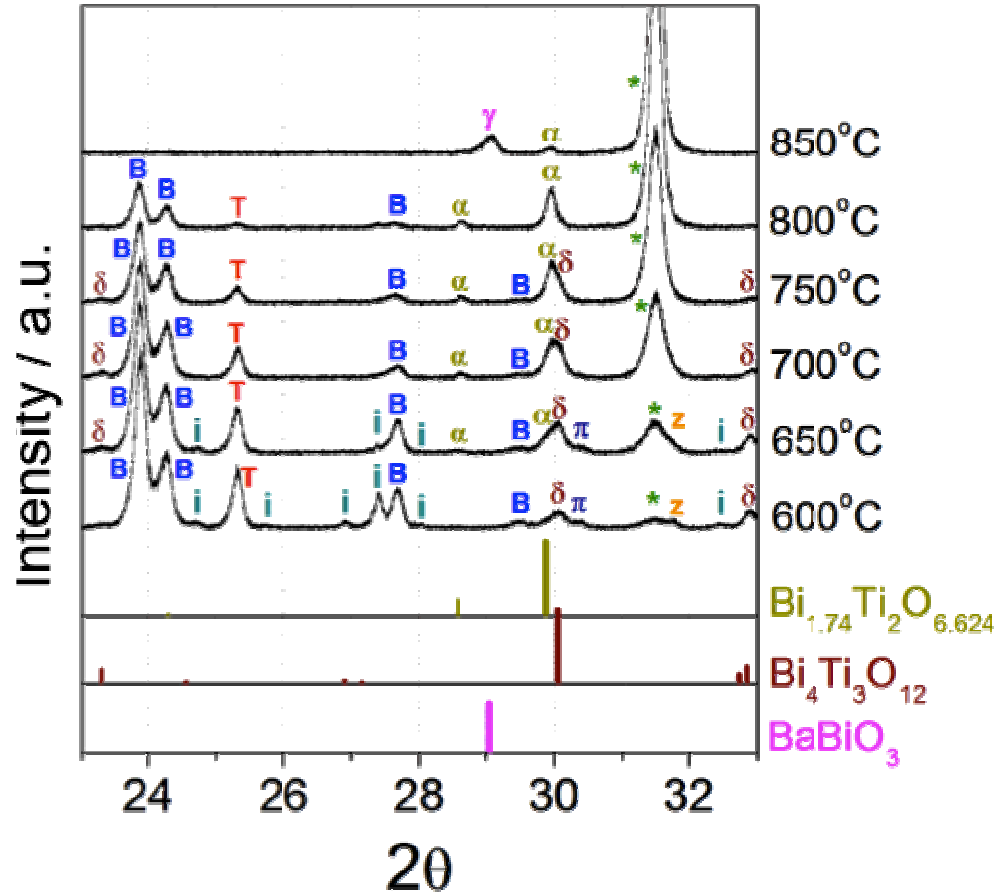
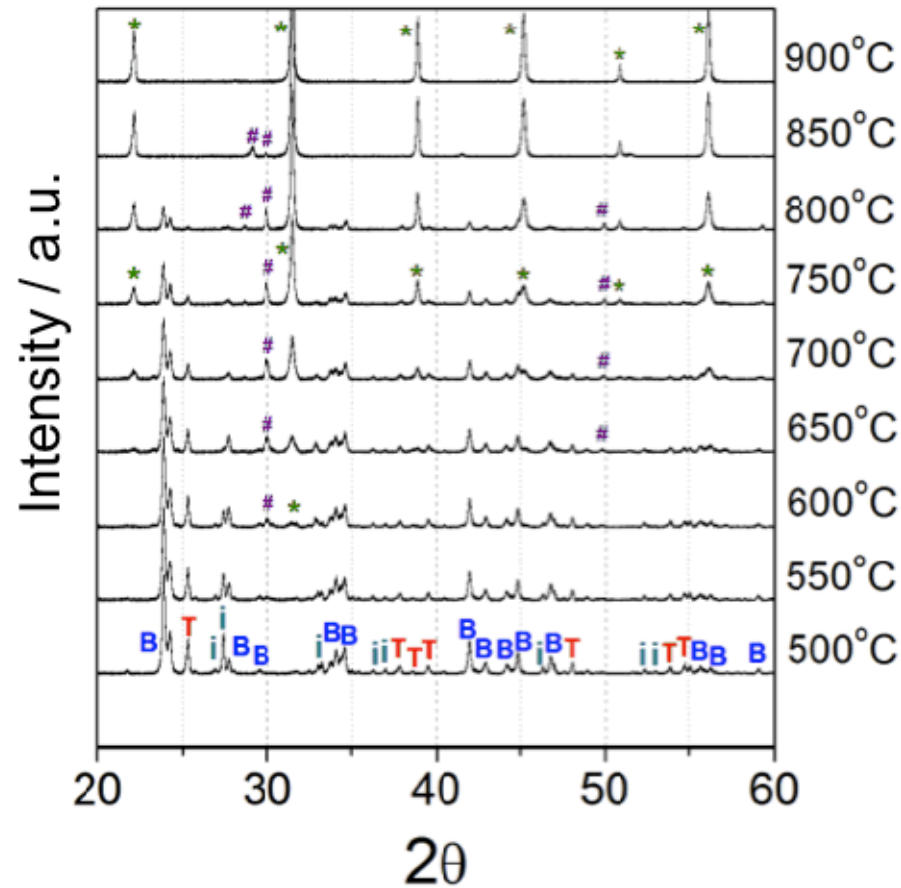
# Calcination: BaTiO<sub>3</sub>



Phase evolution is limited by reaction of BaCO<sub>3</sub>, proceeds via BaTi<sub>2</sub>O<sub>5</sub> and Ba<sub>2</sub>TiO<sub>4</sub> phases



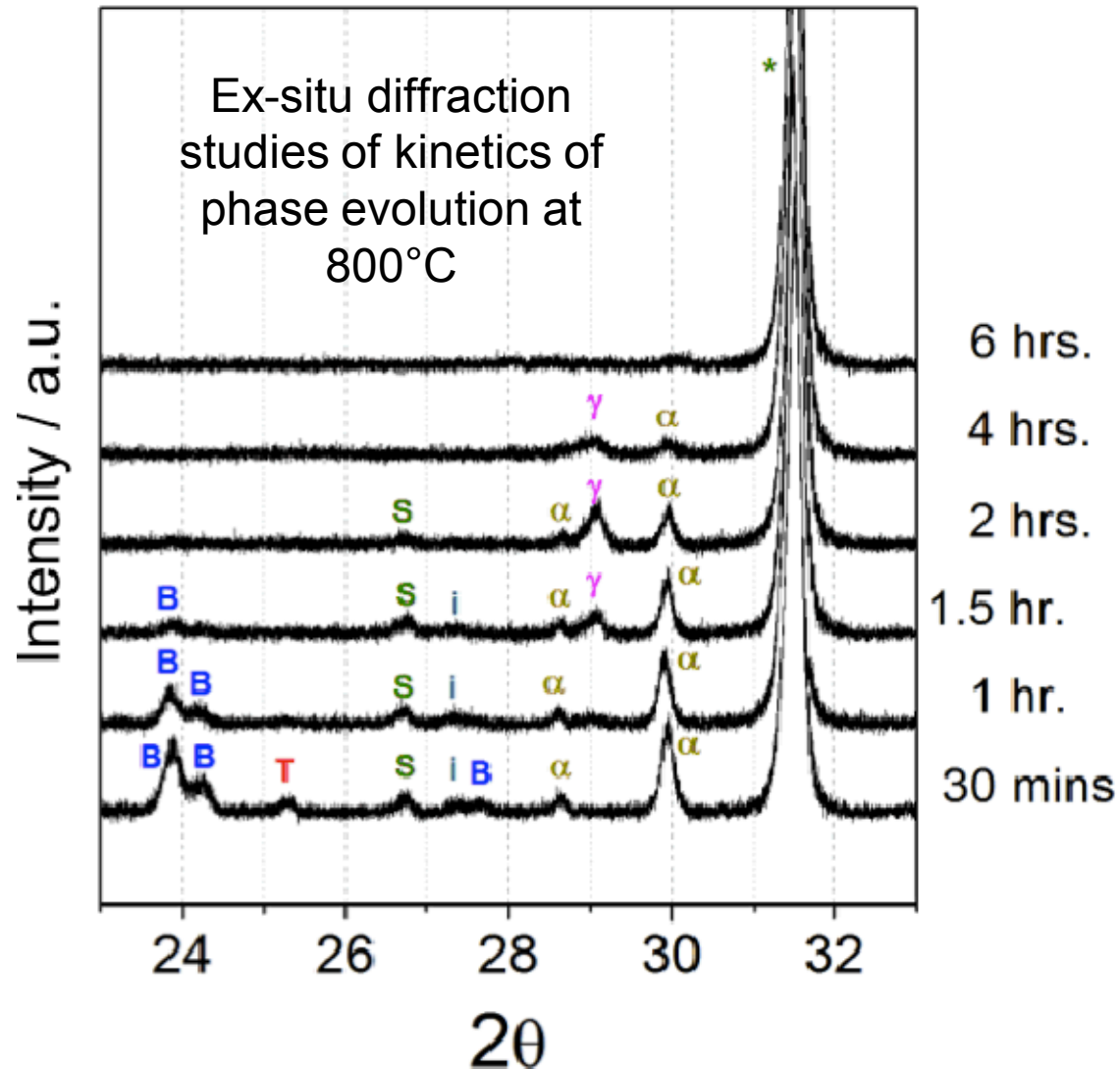
# Calcination: $\text{BaTiO}_3 - \text{Bi}(\text{Zn}_{0.5}\text{Ti}_{0.5})\text{O}_3$



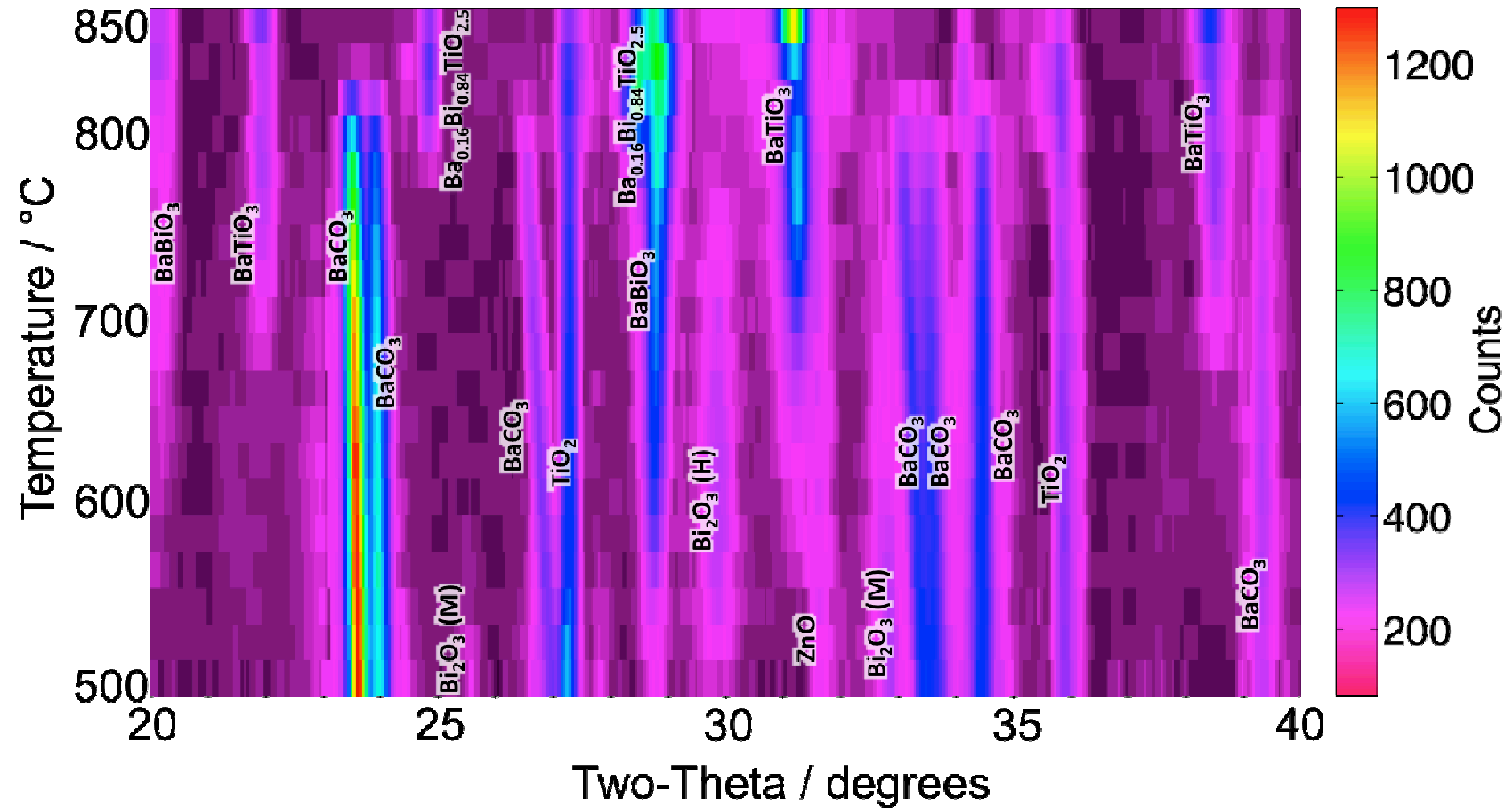
Complex phase evolution with at least 3 intermediate phases, but BZT additions result in single-phase perovskite 200°C lower than pure  $\text{BaTiO}_3$

# Calcination: $\text{BaTiO}_3 - \text{Bi}(\text{Zn}_{0.5}\text{Ti}_{0.5})\text{O}_3$

Phase development via mixed-Bi-valence  $\text{BaBiO}_3$  phase (likely with significant substitution on both sites...) appears to be critically important for microstructure development.

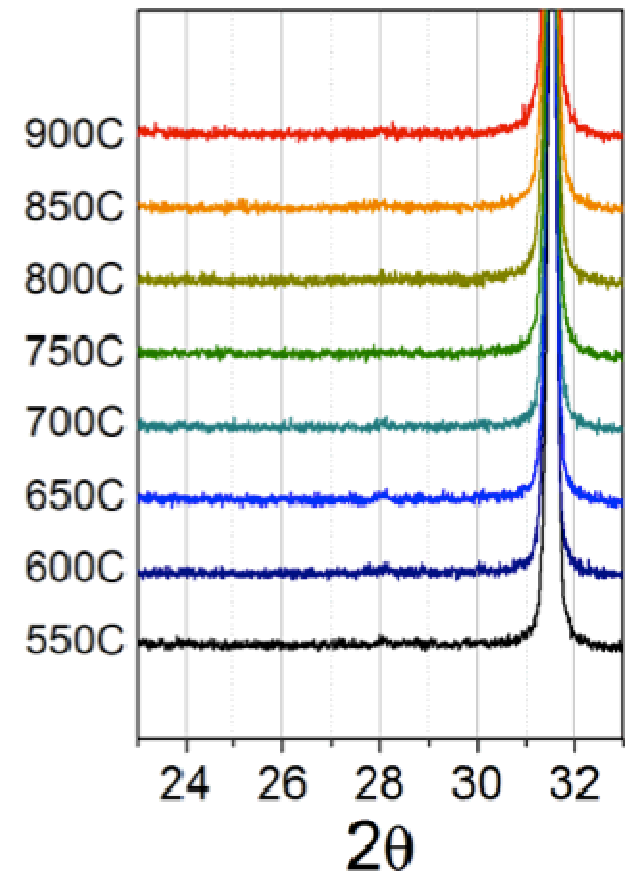
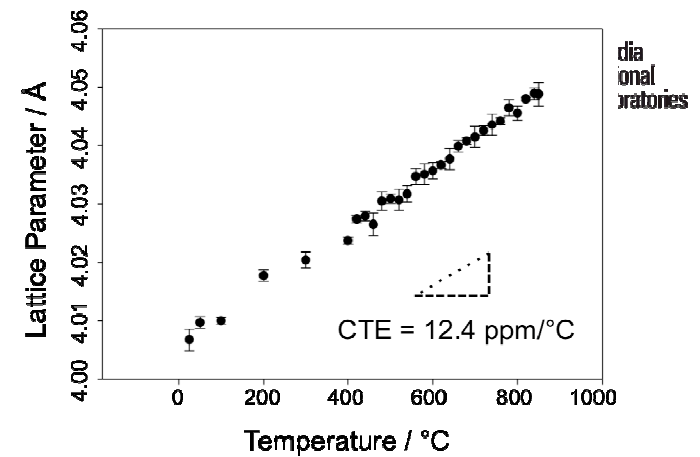
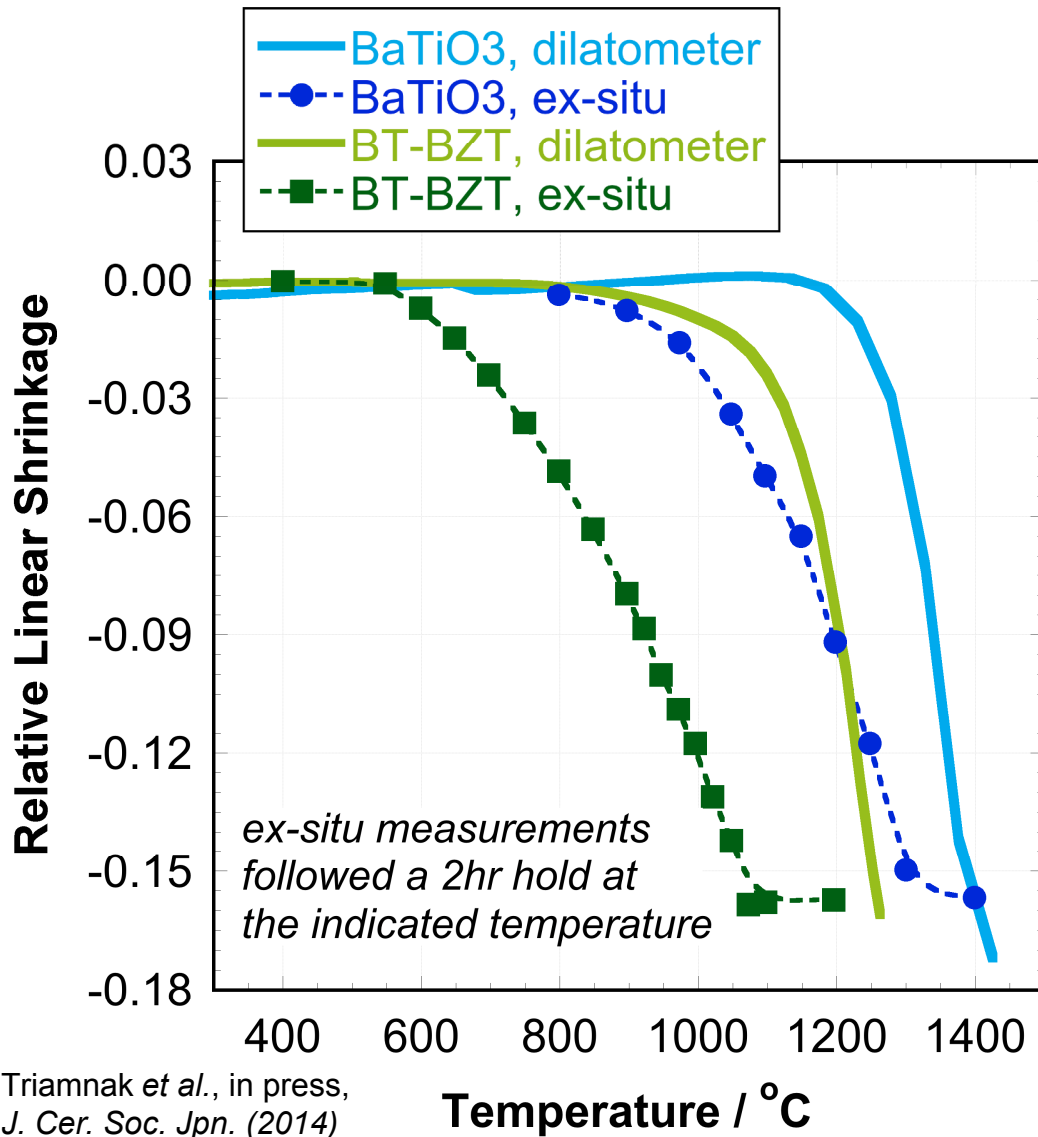


# Calcination



Triamnak *et al.*, in press,  
*J. Cer. Soc. Jpn.* (2014)

# Sintering

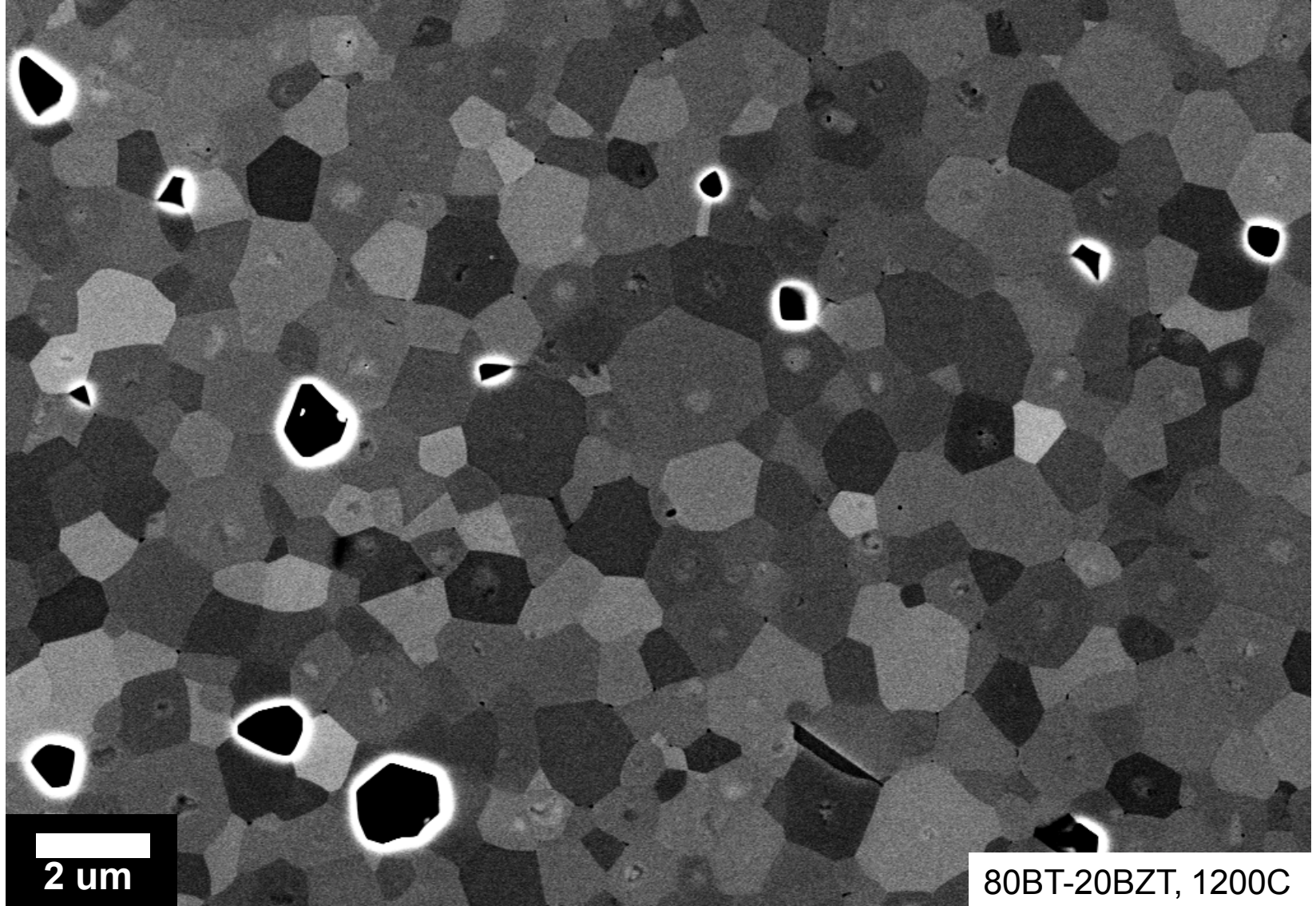


# Thermally Etched Surface

 The image cannot be displayed. Your computer may not have enough memory to open the image, or the image may have been corrupted. Restart your computer, and then open the file again. If the red x still appears, you may have to delete the image and then insert it again.



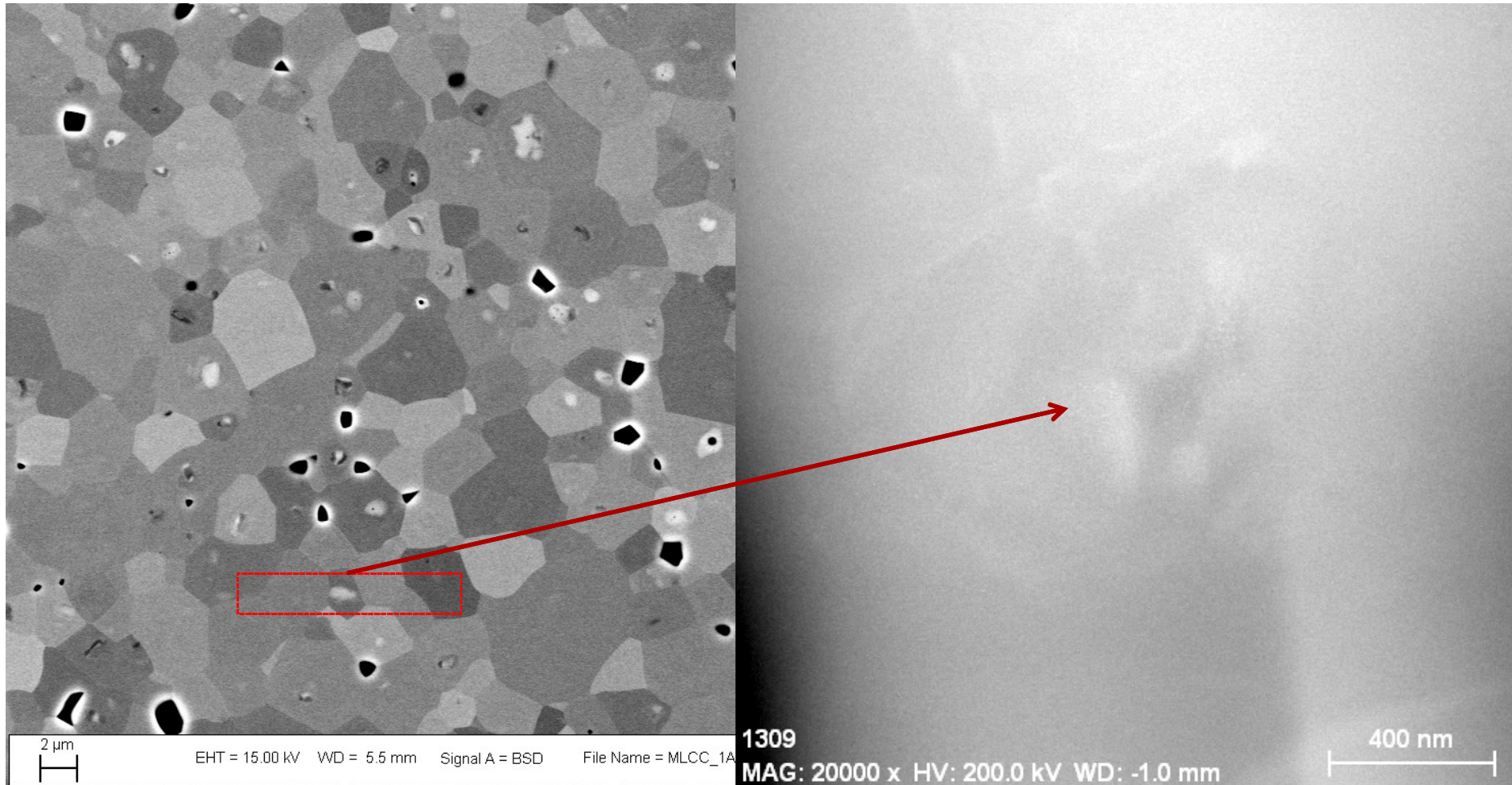
# Well-Polished Section, Channeling



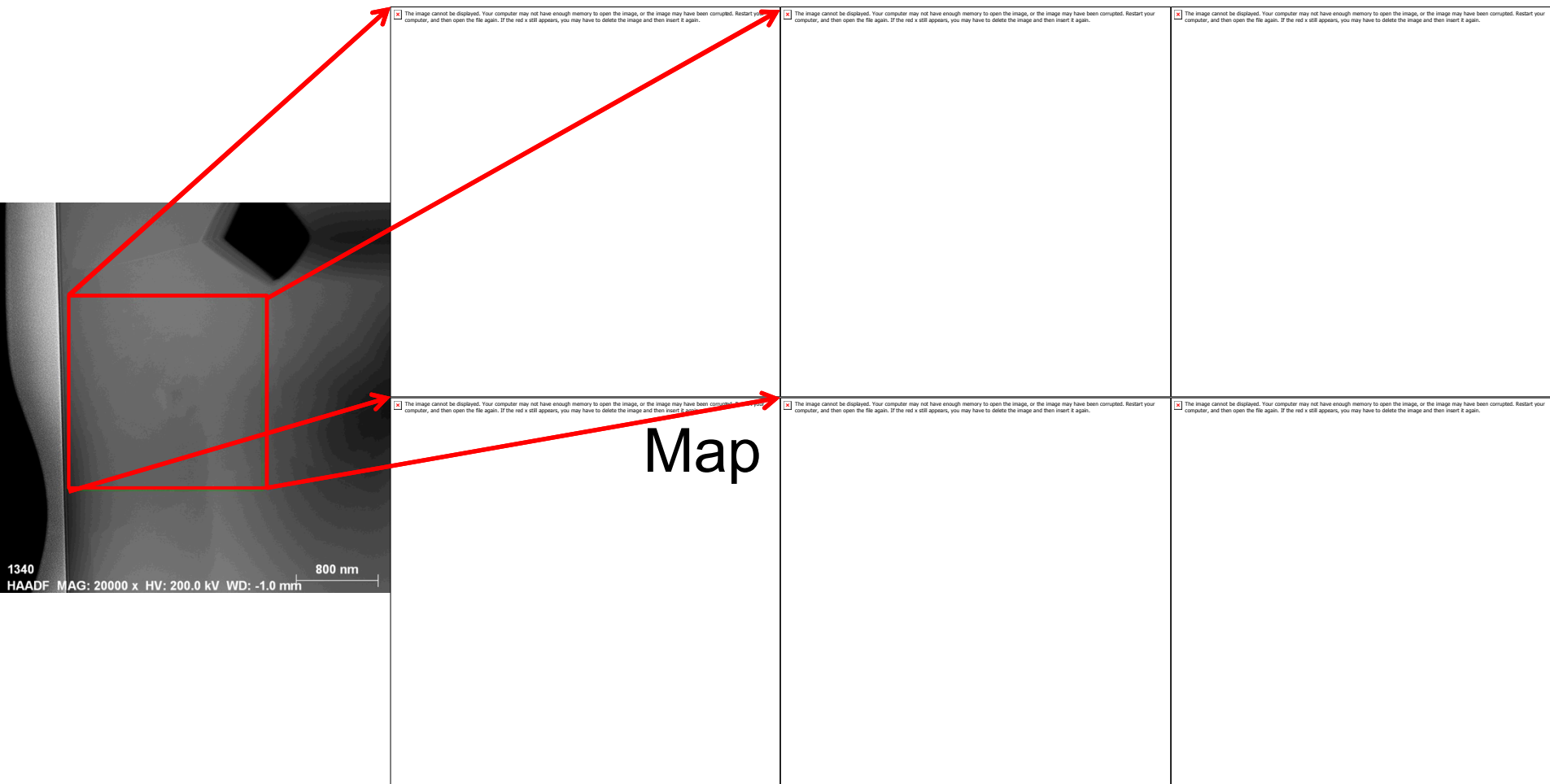
# Compositional Variation

SEM

HAADF TEM



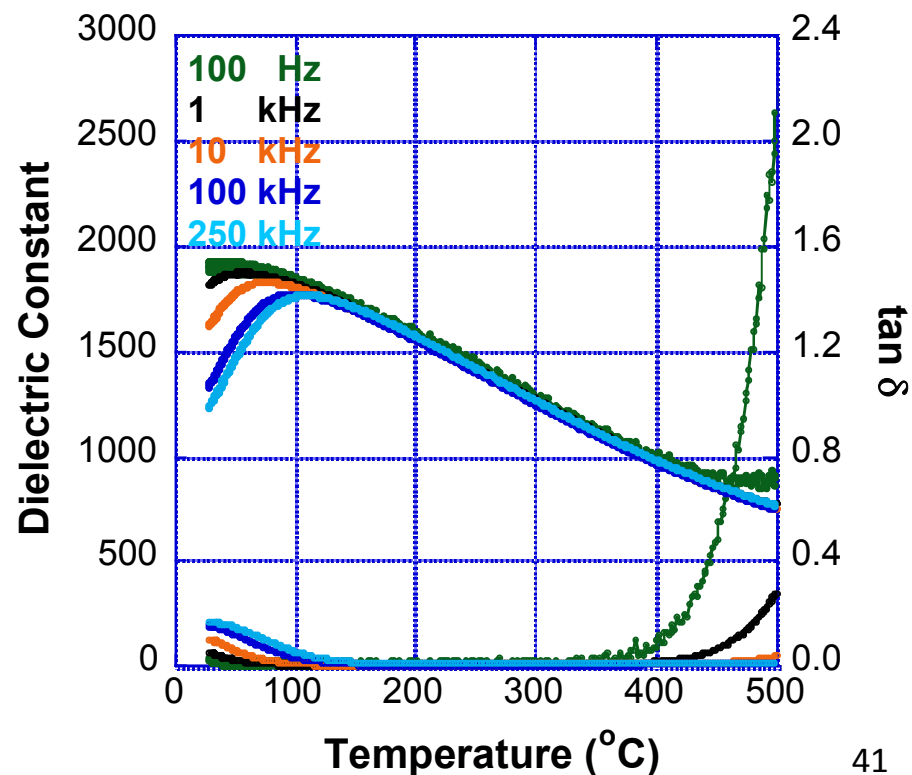
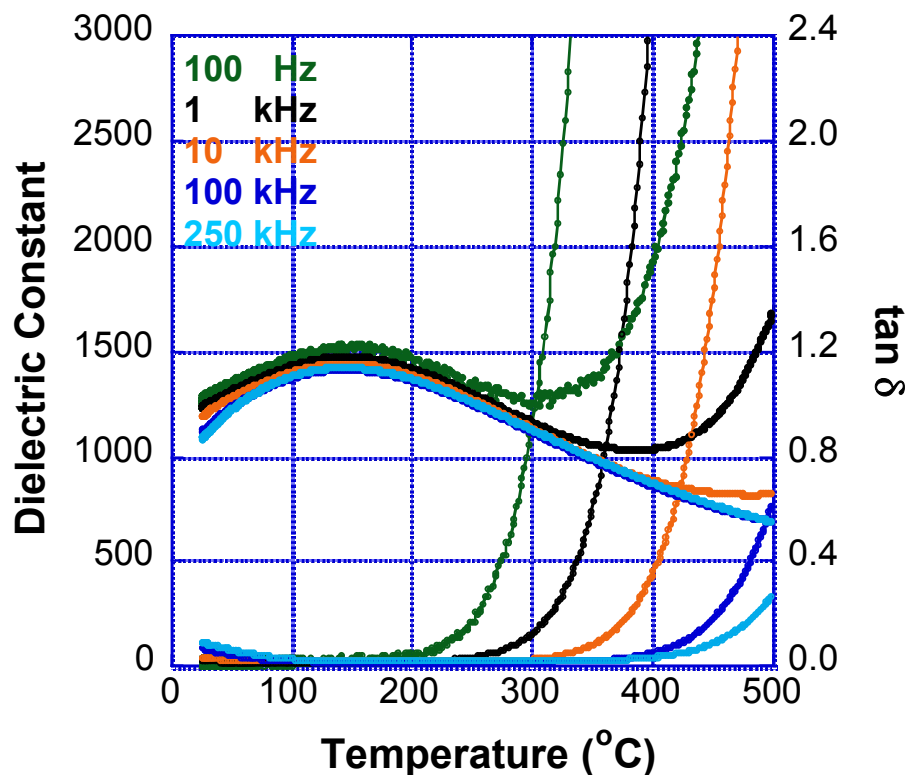
# Bi and Zn Co-segregation



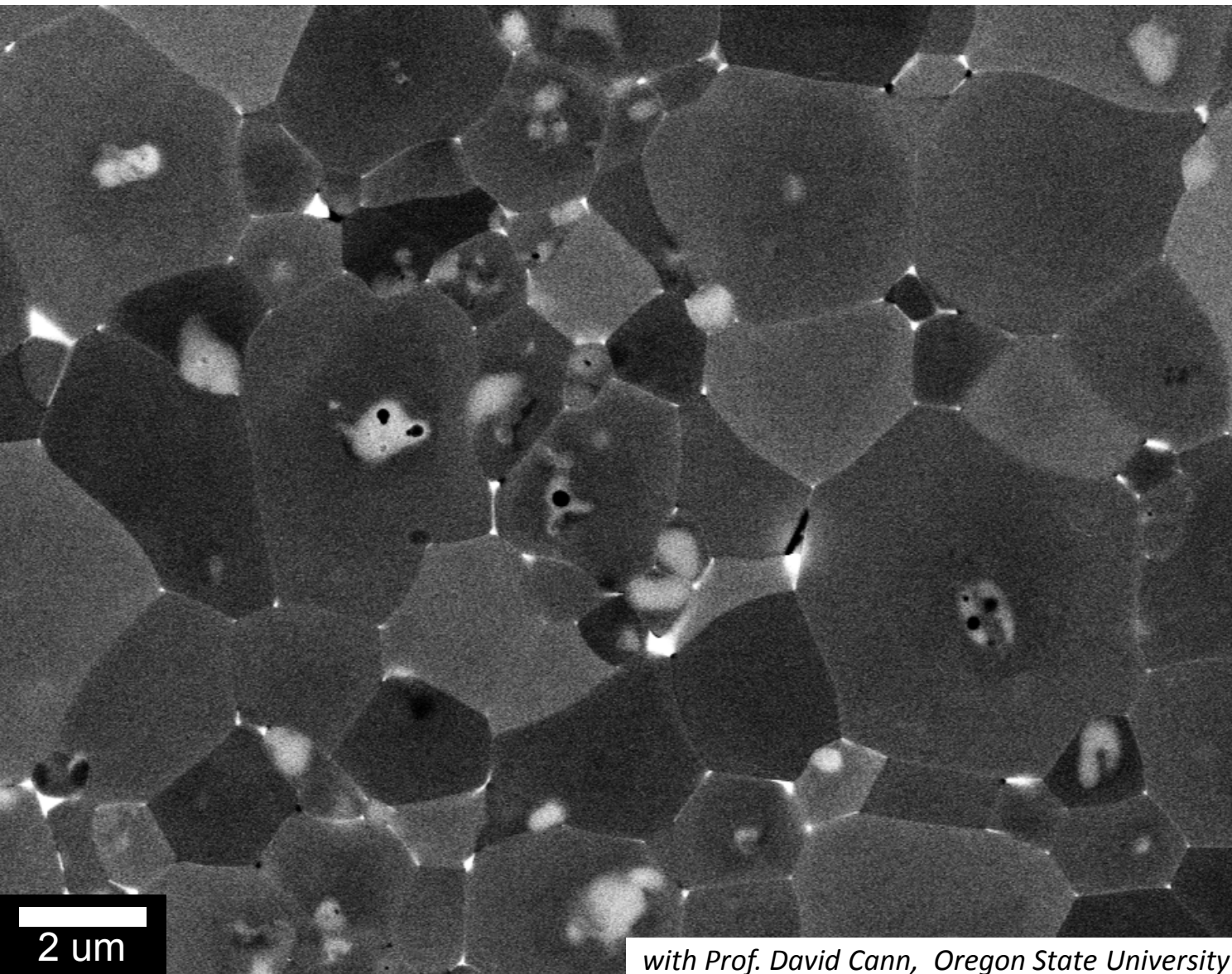


# Defect Chemistry Studies

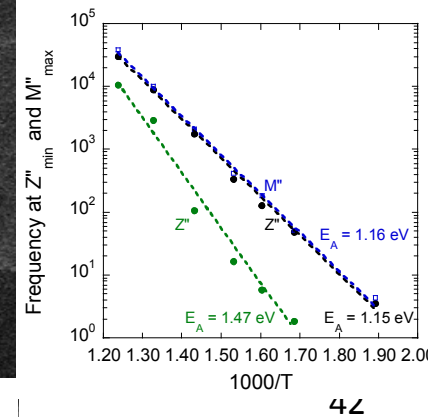
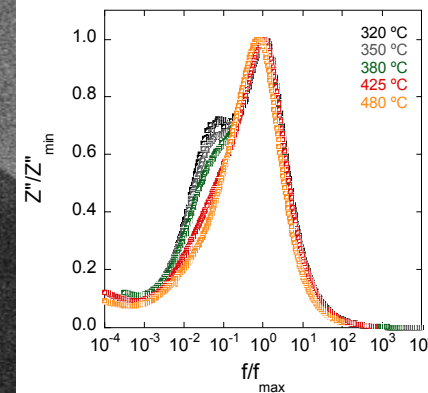
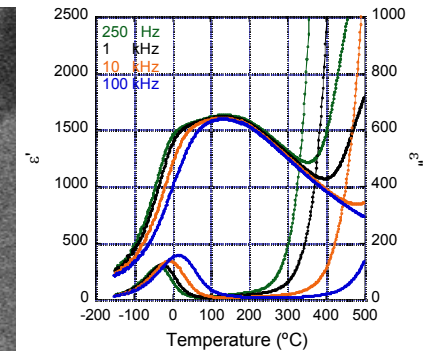
Acceptor (up to 2mol%)	Donor
Low-purity TiO <sub>2</sub> precursor	Ba deficient (compensating)
Ba excess (sub for Bi)	Bi excess (sub for Ba)
Zn excess (sub for Ti)	Ti excess (sub for Zn)
Mn sub for Ti	Nb sub for Ti



# Acceptor Doped

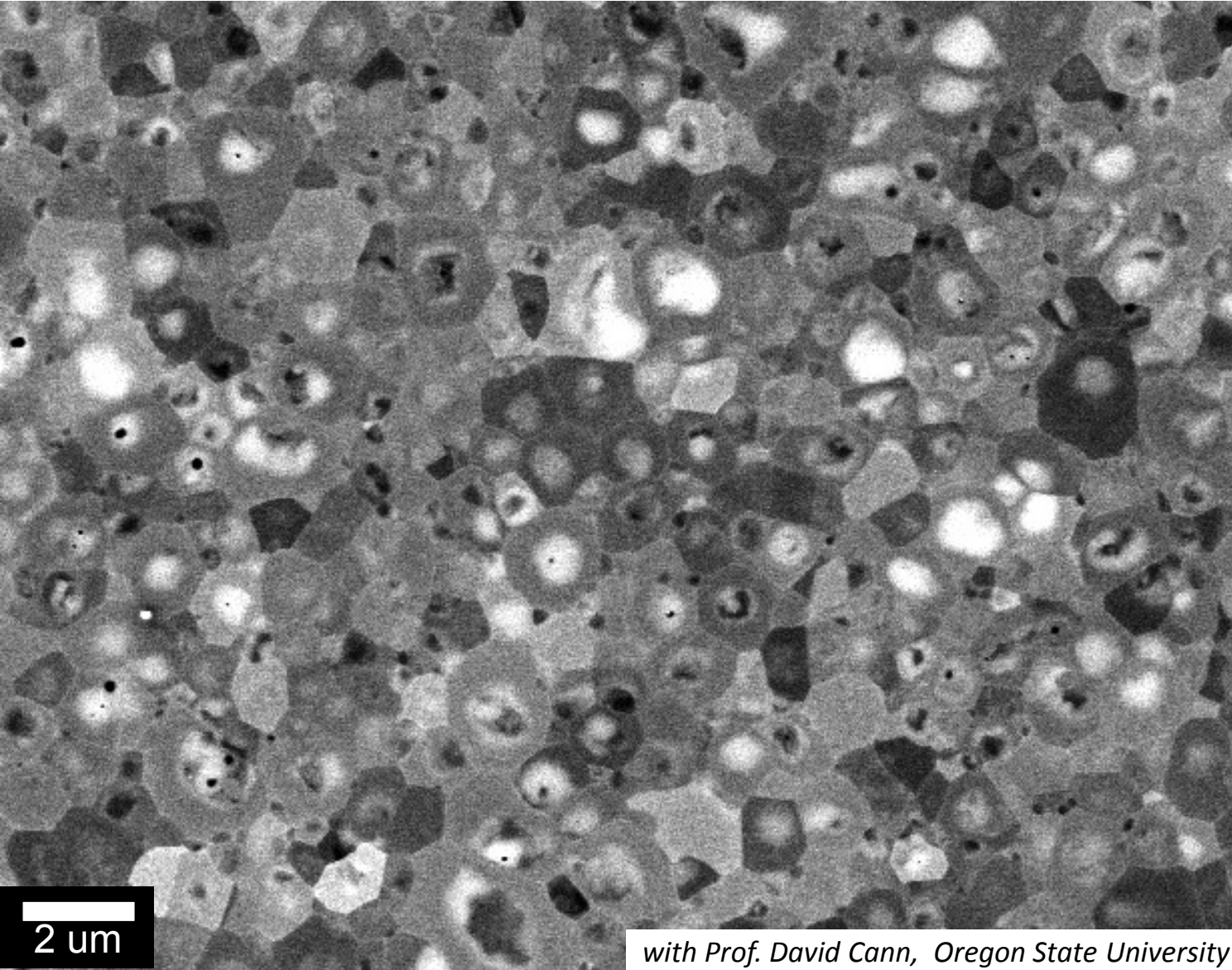


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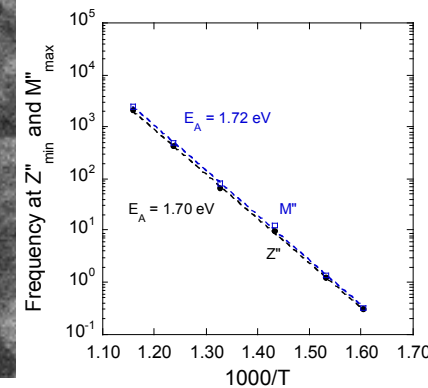
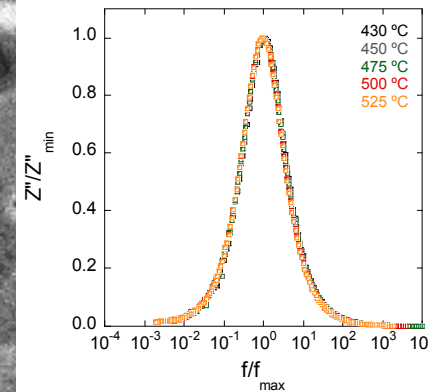
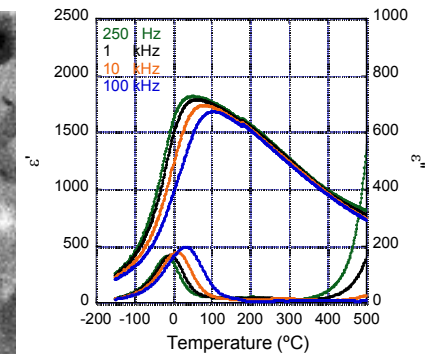




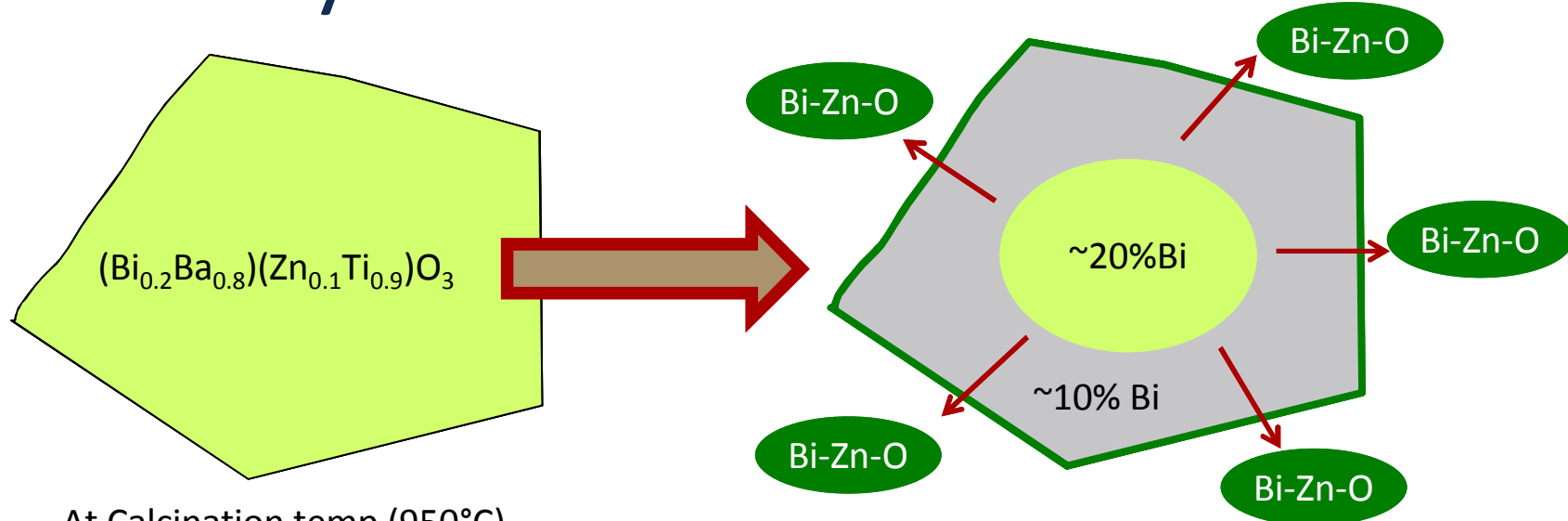
# Donor Doped



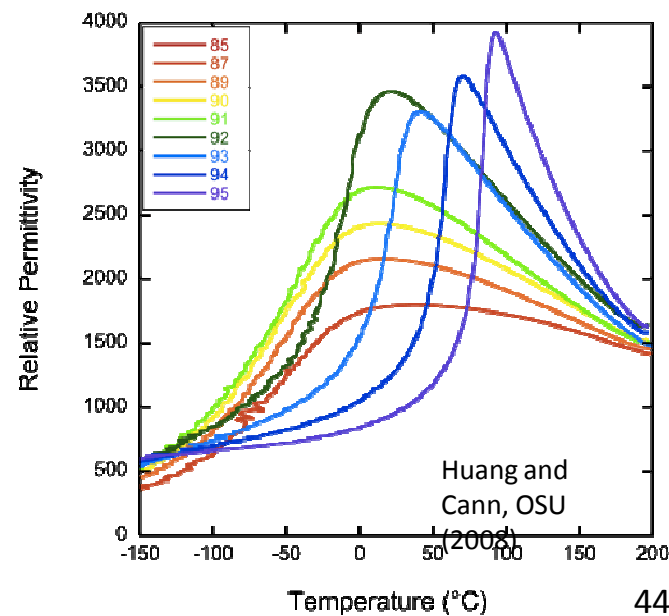
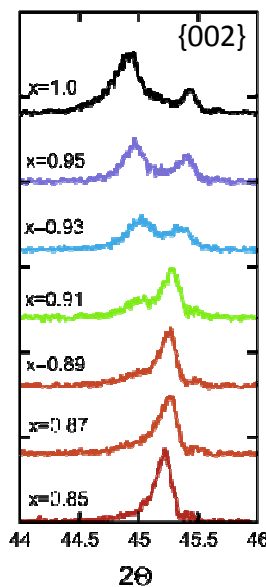
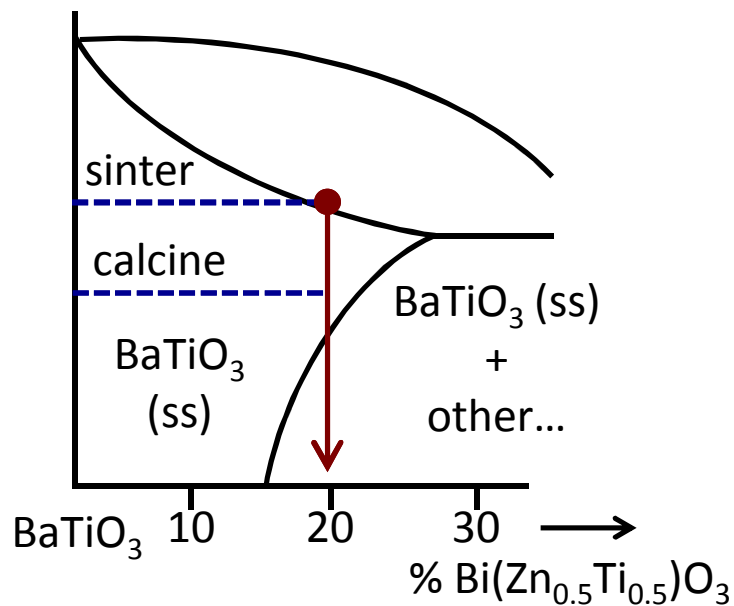
with Prof. David Cann, Oregon State University



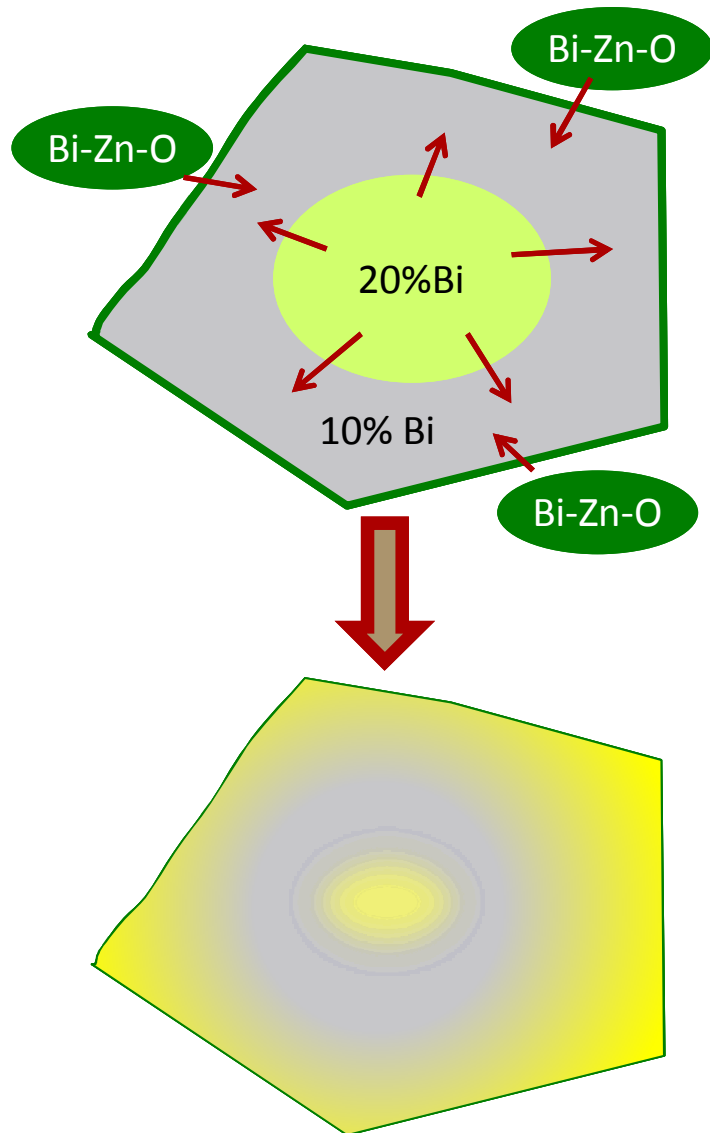
# Solubility Limitation?



At Calcination temp ( $950^\circ\text{C}$ )



# Sintering: Donor doped



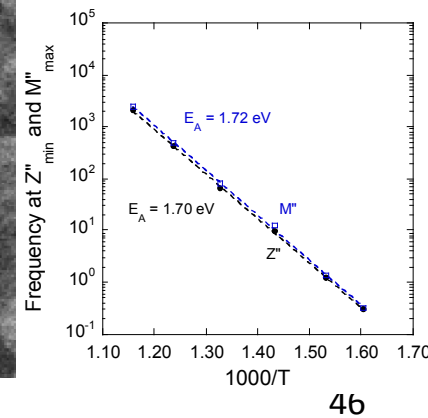
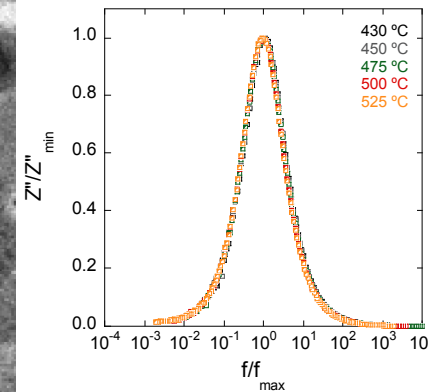
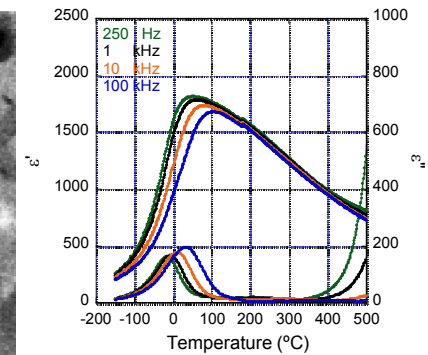
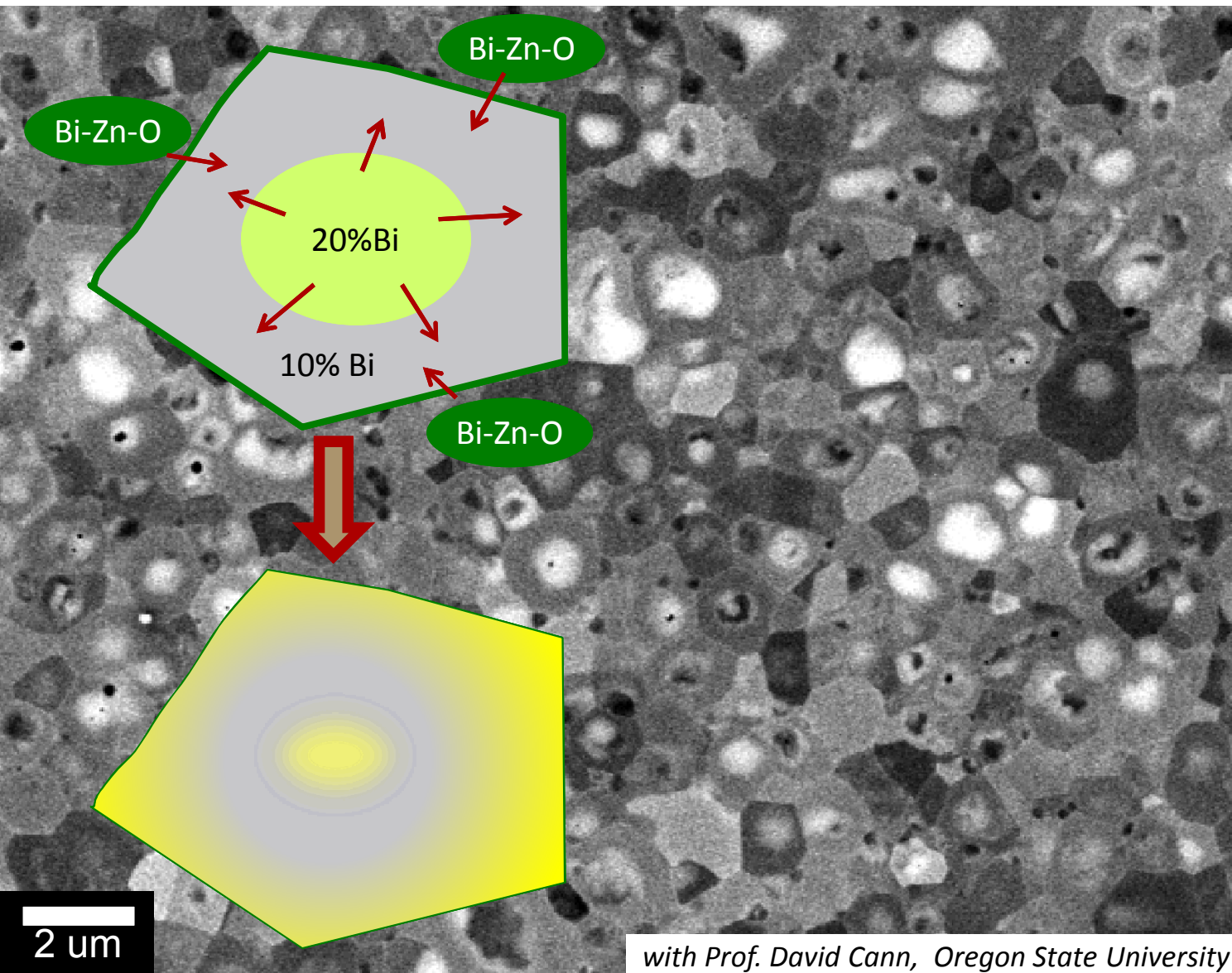
## Donor Doped: Cation Vacancies Dominate

- Bi diffuses in from g.b., out from core
- Diffusion **assisted** by cation vacancies

## Resultant Microstructure

- Relatively homogenous
- Bi-gradient with diffuse boundary due to ample Bi diffusion
- Electrically homogeneous microstructure (single relaxation)

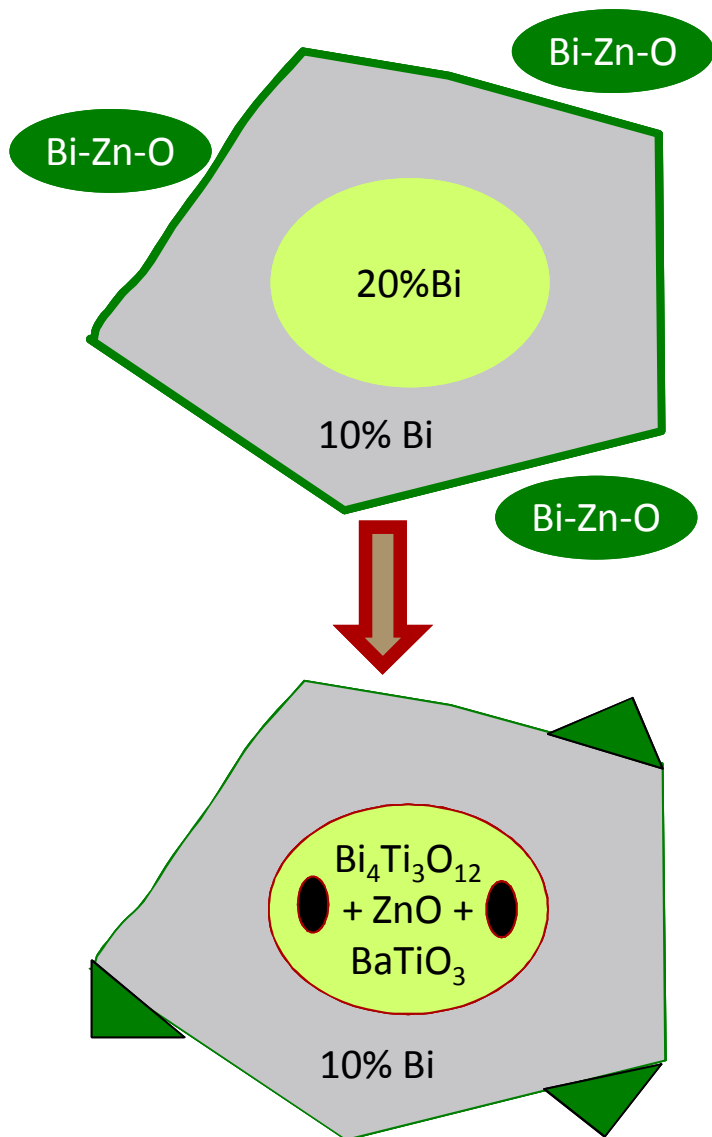
# Sintering: Donor doped



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# Sintering: Acceptor doped



## Acceptor Doped: Oxygen Vacancies Dominate

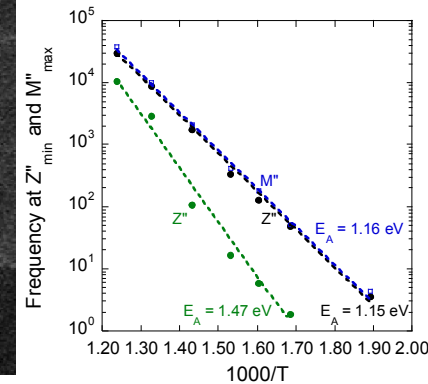
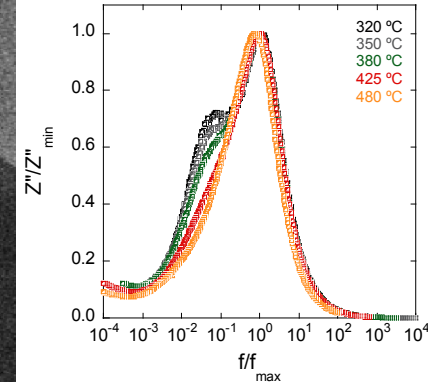
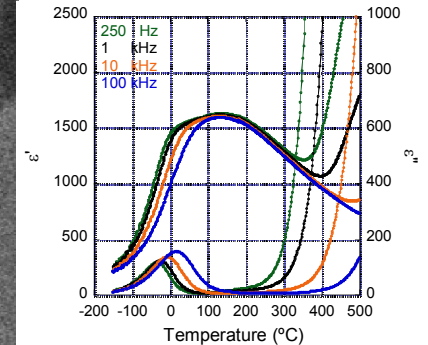
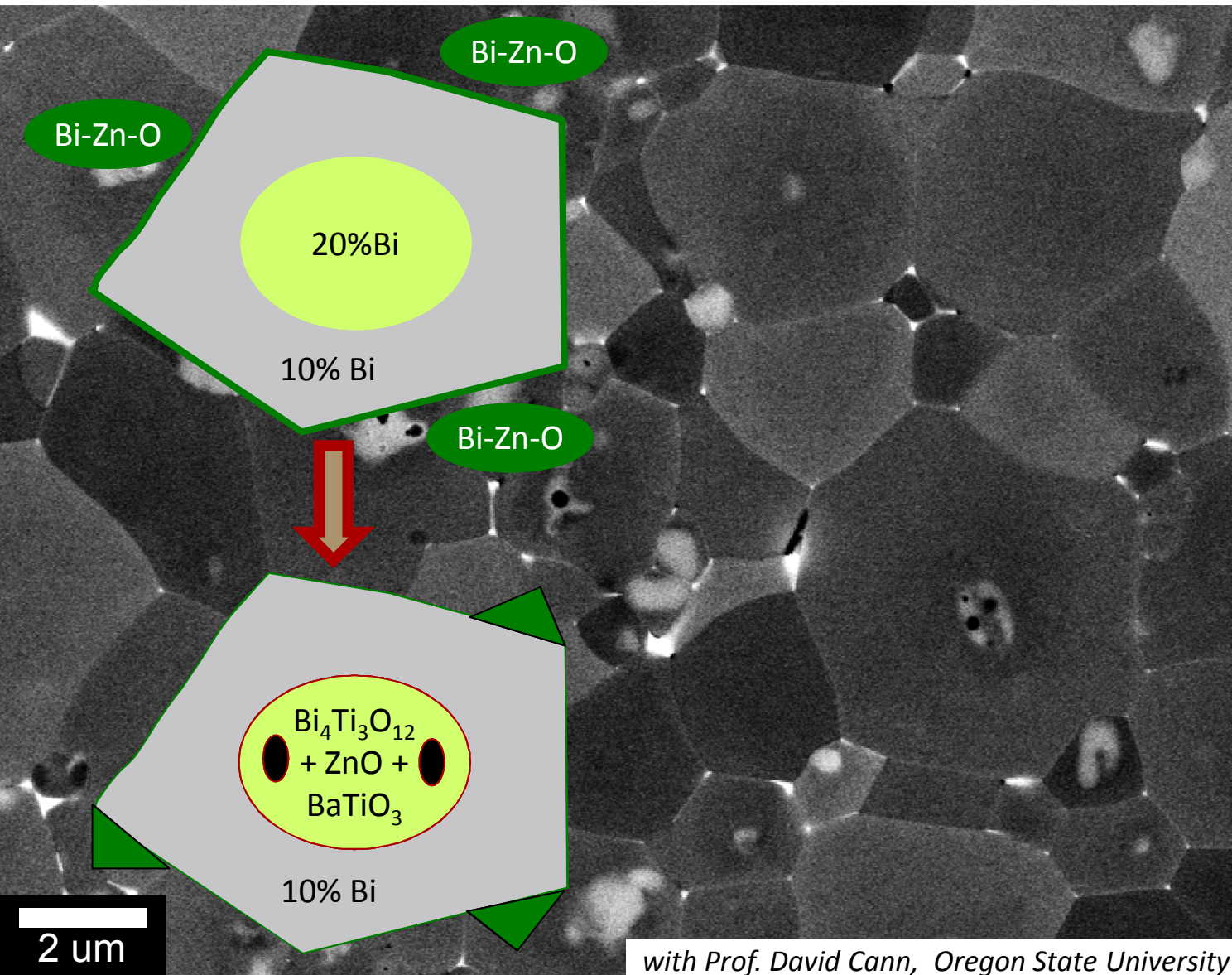
- Bi diffuses in from g.b., out from core
- Diffusion **inhibited** by lack of cation vacancies
- Bi-rich phase at triple points
- Bi-rich cores decompose into equilibrium phases:  $\text{Bi}_4\text{Ti}_3\text{O}_{12} + \text{ZnO} + \text{Ba-Ti-O}$

## Resultant Microstructure

- High-Z phase at triple points
- Well defined core boundary
- Low-Z precipitates in core region
- Electrically heterogeneous microstructure (two relaxations)

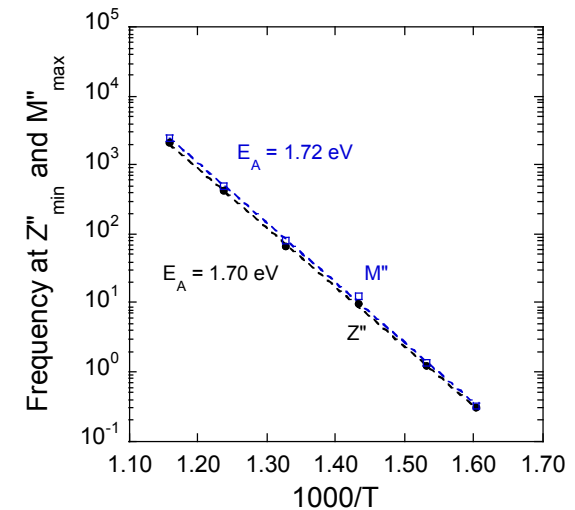
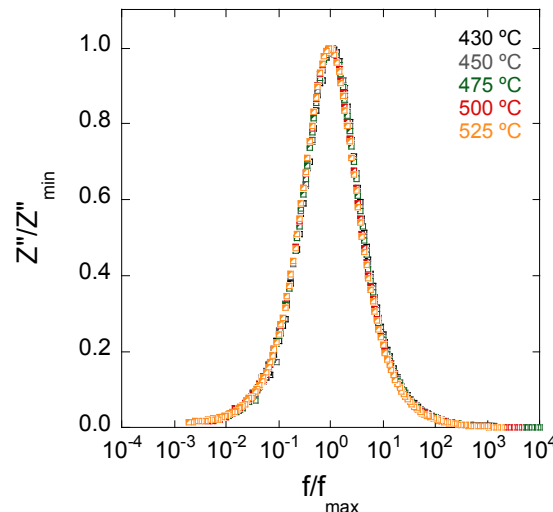
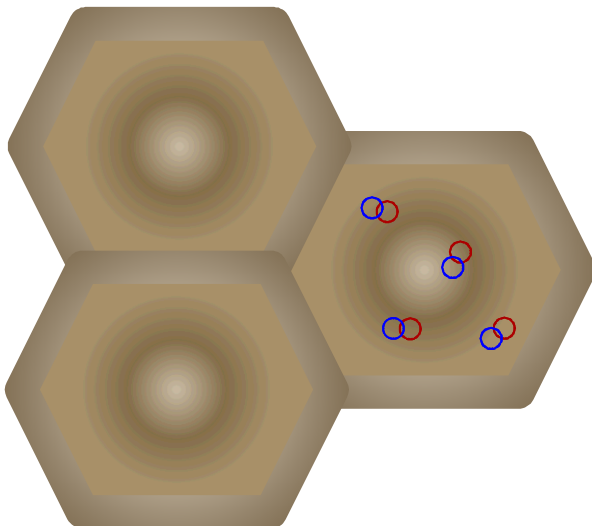
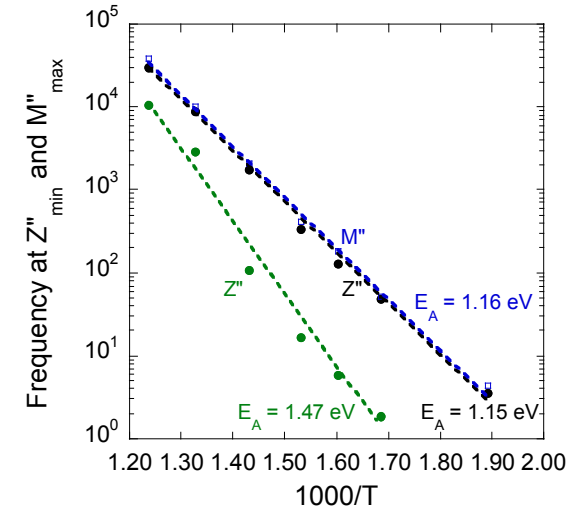
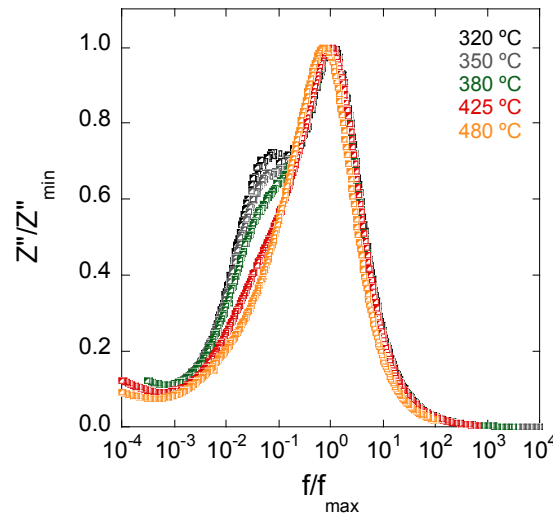
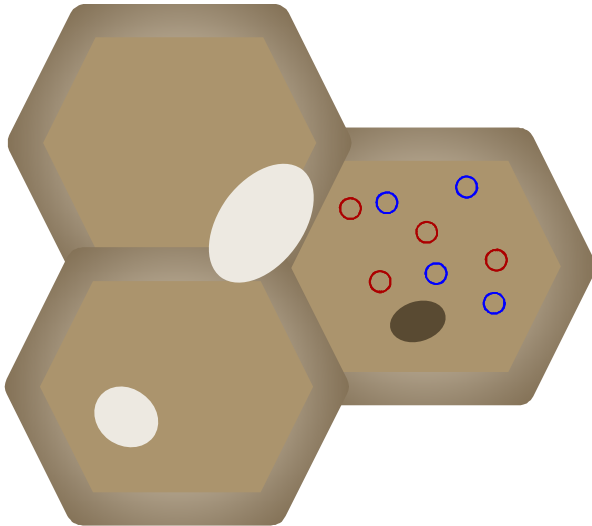


# Sintering: Acceptor doped



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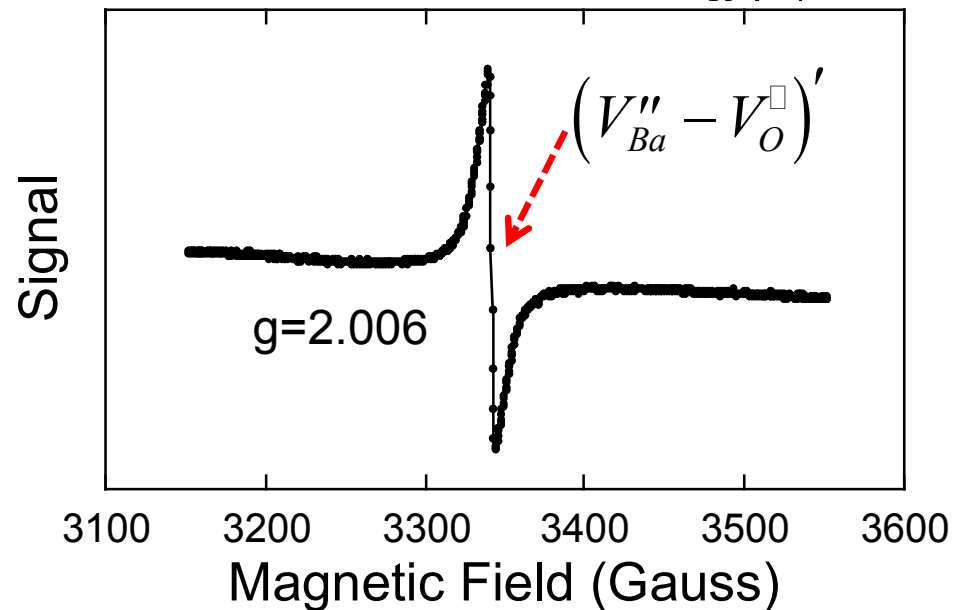
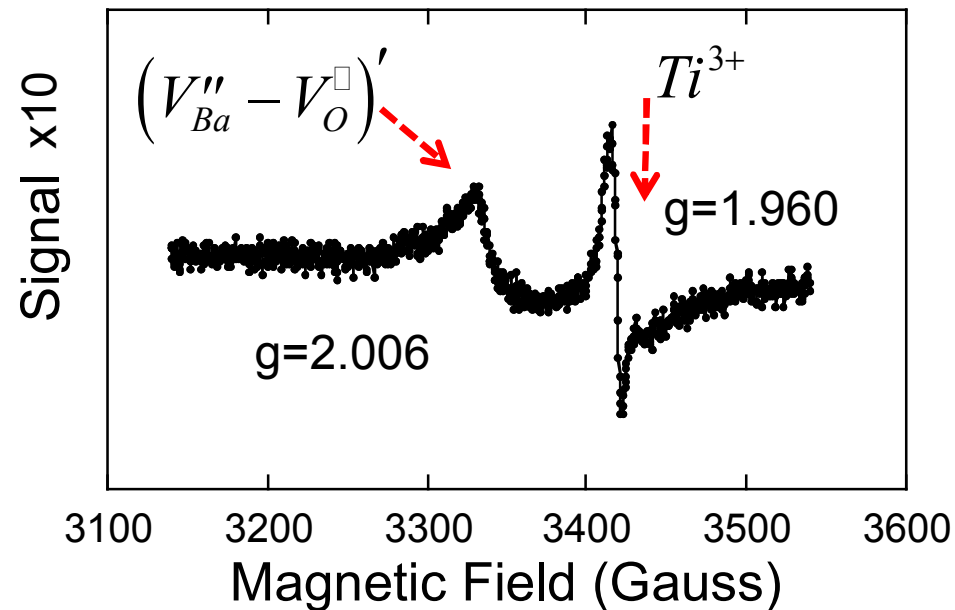
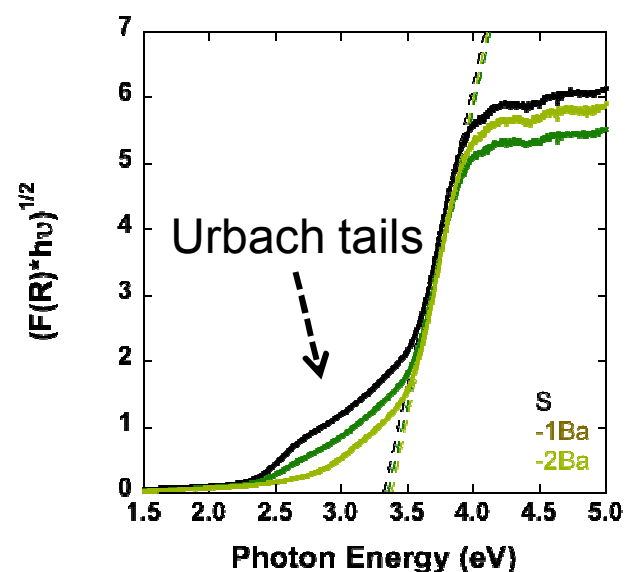
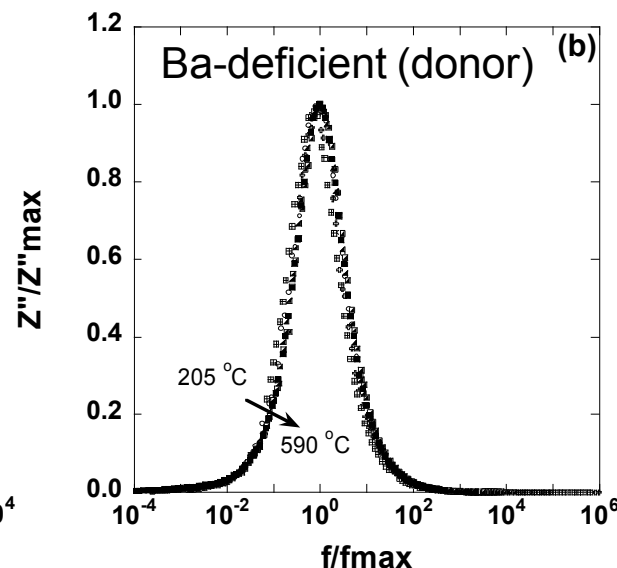
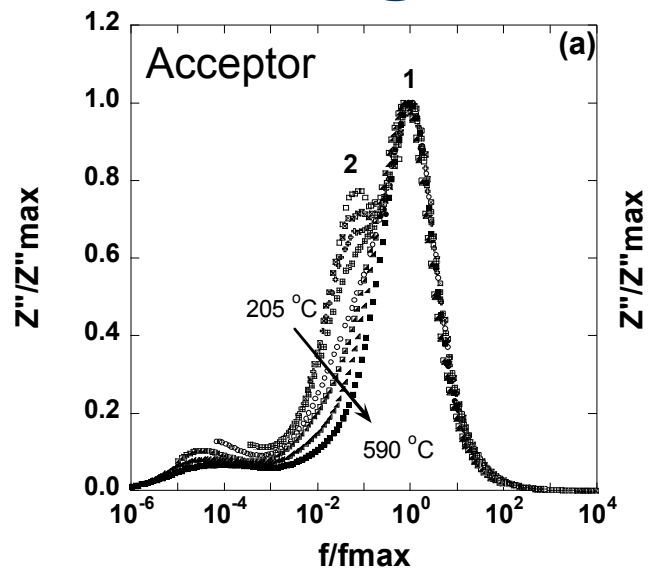
# Microscale Heterogeneity



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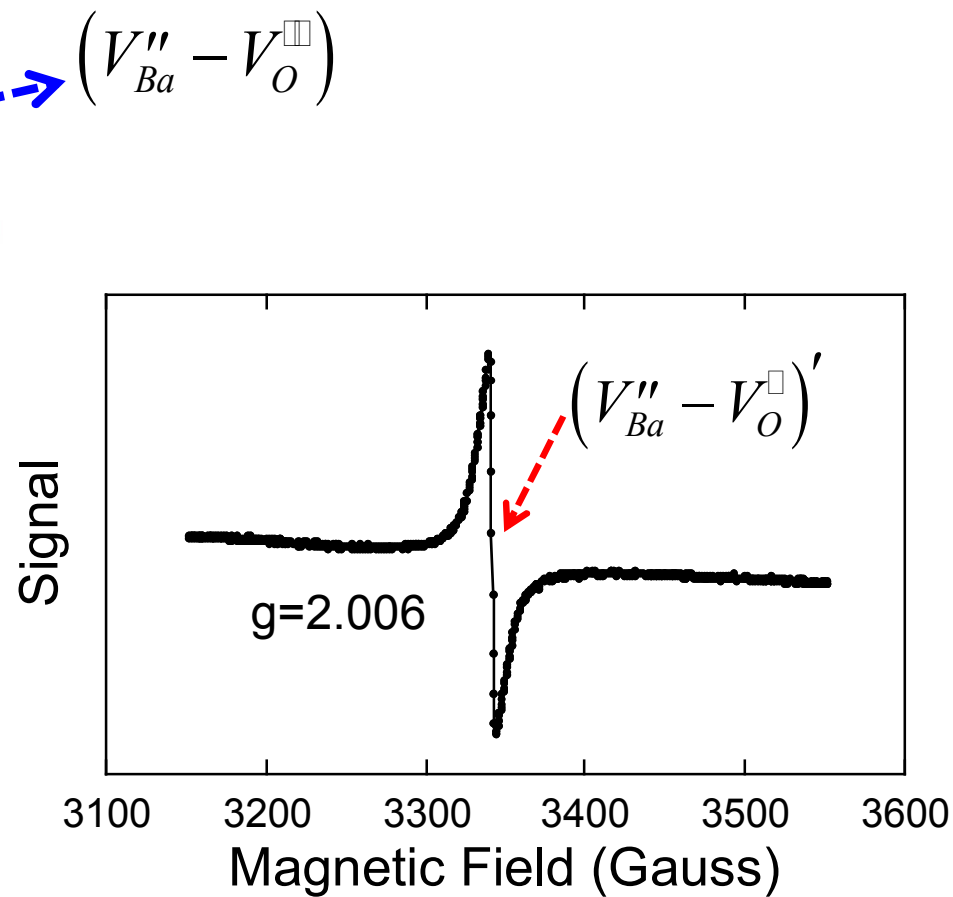
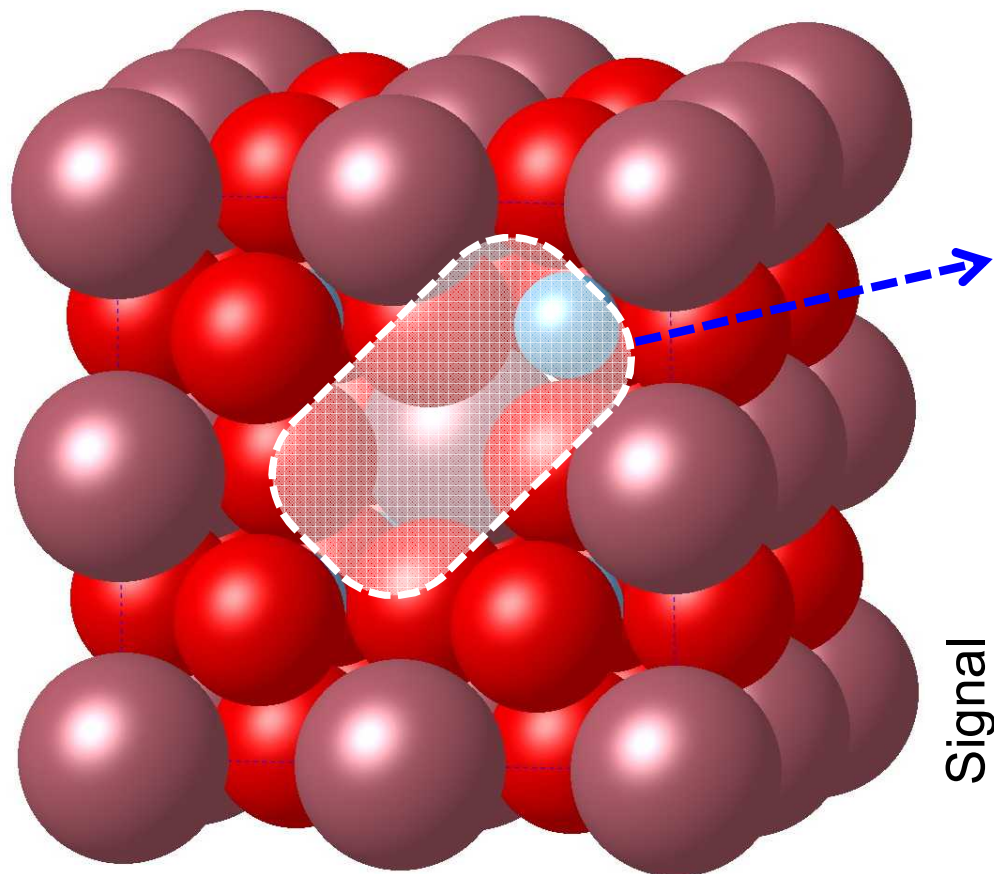
# Probing Defects

Raengthon et al., *Appl. Phys. Lett.*, **101** (2012) 112904.



# Probing Defects

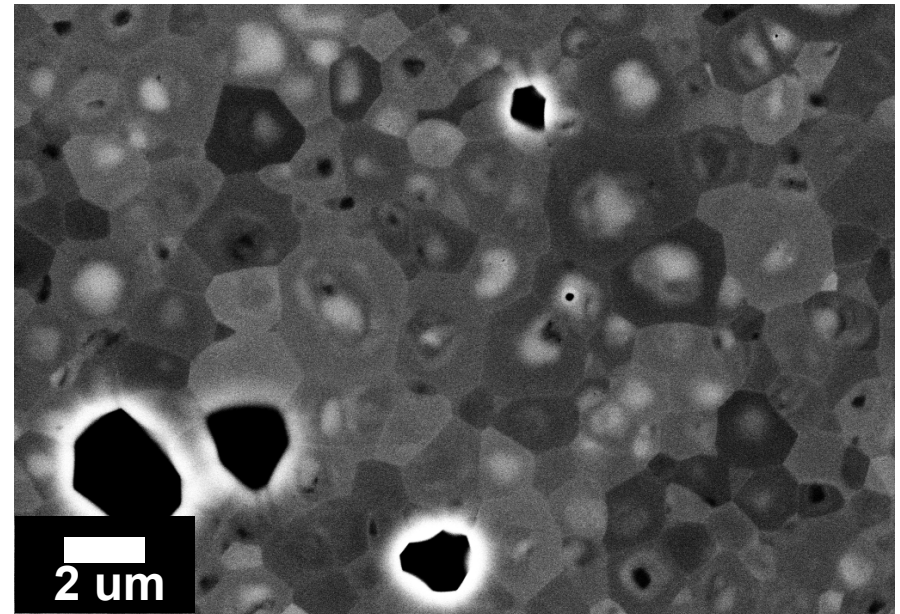
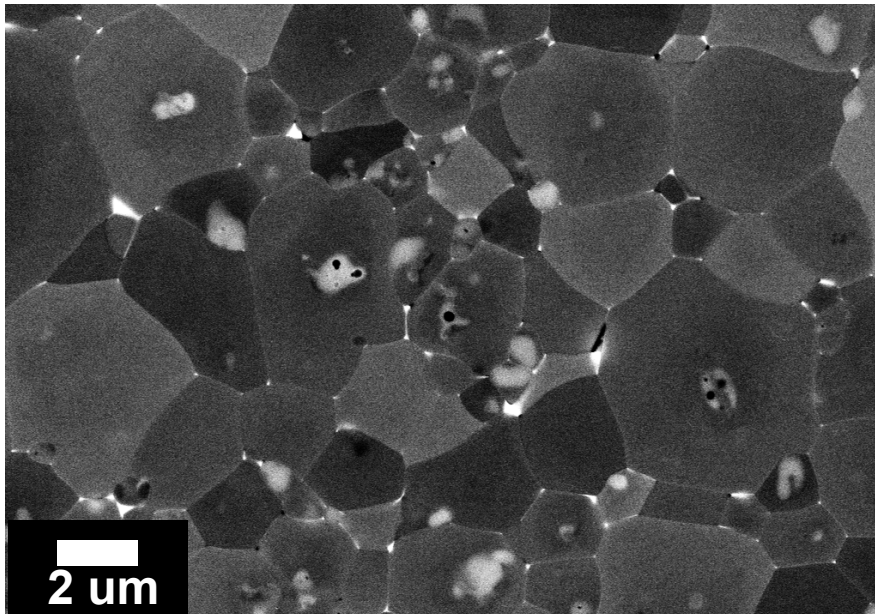
Raengthon et al., *Appl. Phys. Lett.*, **101** (2012) 112904.





# Summary

- Complex phase evolution and potential liquid phase(s) enable reduced temperature processing
- Electrical response(s) of weakly-coupled relaxor systems tied to multi-scale chemical heterogeneities
- Tightly bound defects enable high resistivity





# Extra Slides Follow