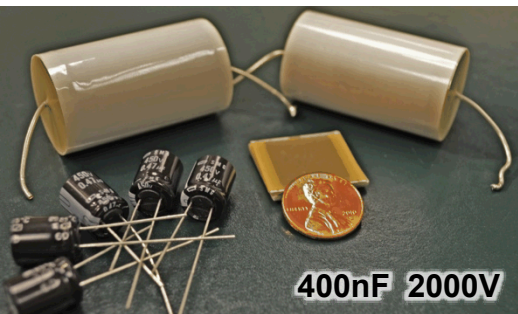
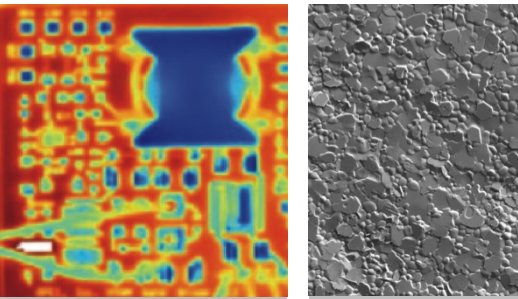


Ceramic Capacitors based on Temperature and Voltage-Stable Relaxor Dielectrics



Geoff Brennecka, Harlan Brown-Shaklee,
John Borchardt, and Mia Blea-Kirby

Sandia National Laboratories

Natthaphon Raengthon and David Cann

Oregon State University

Kevin Ring

New Mexico Tech



Sandia
National
Laboratories

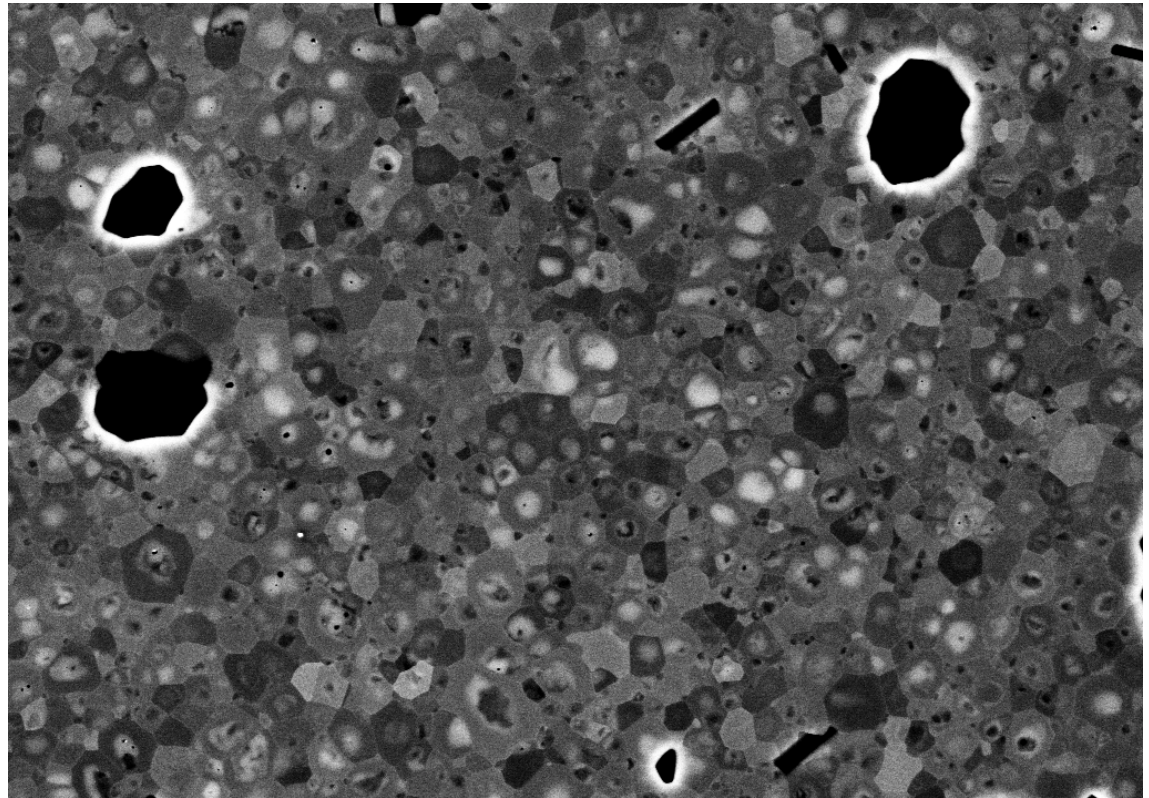
*Exceptional
service
in the
national
interest*



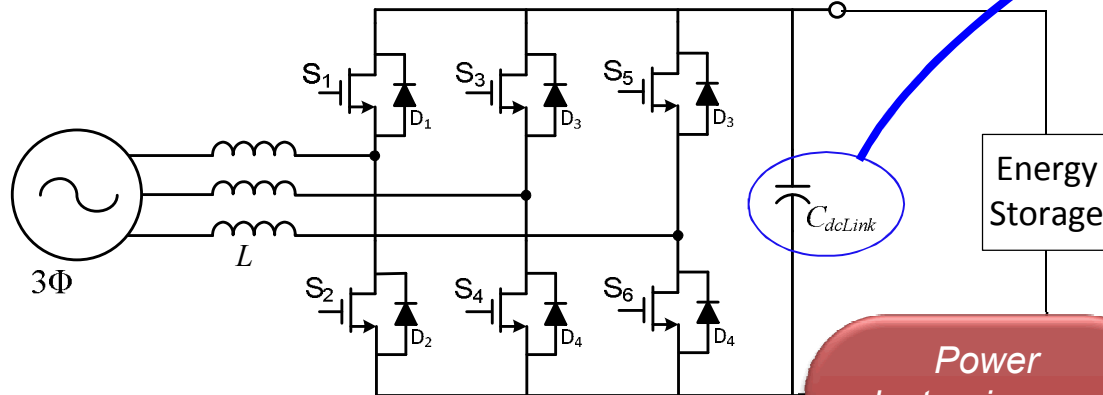
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 - Stan Atcitty
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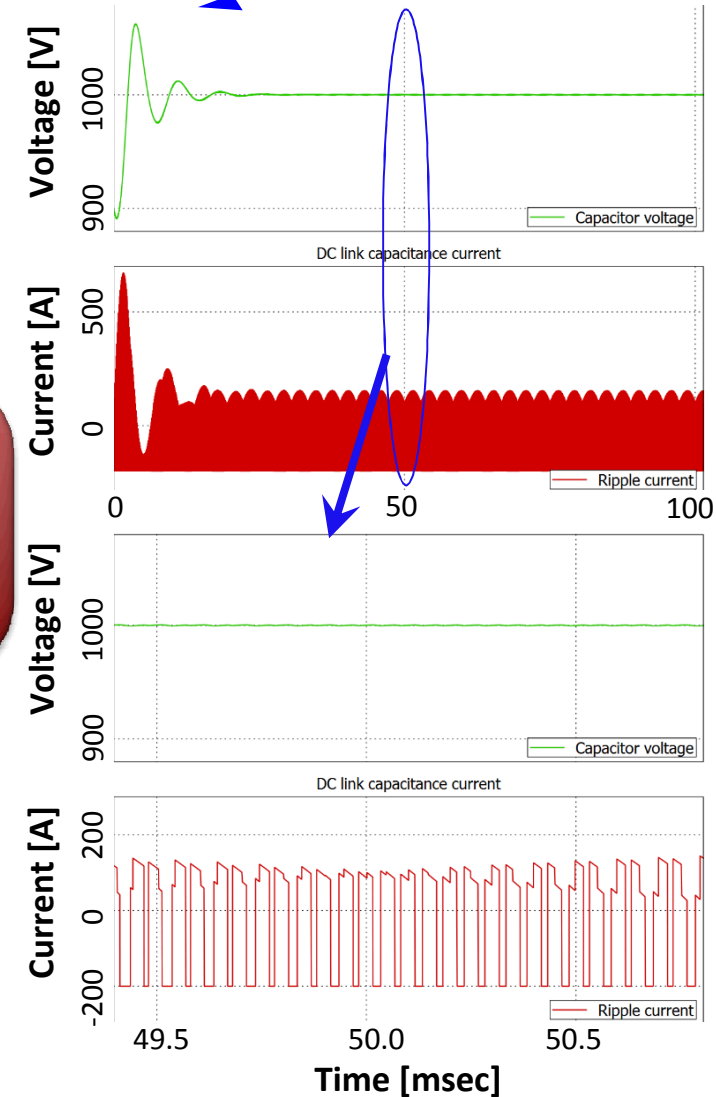


Application Space



Power electronics need high operating temperature, low ESR, high resistivity

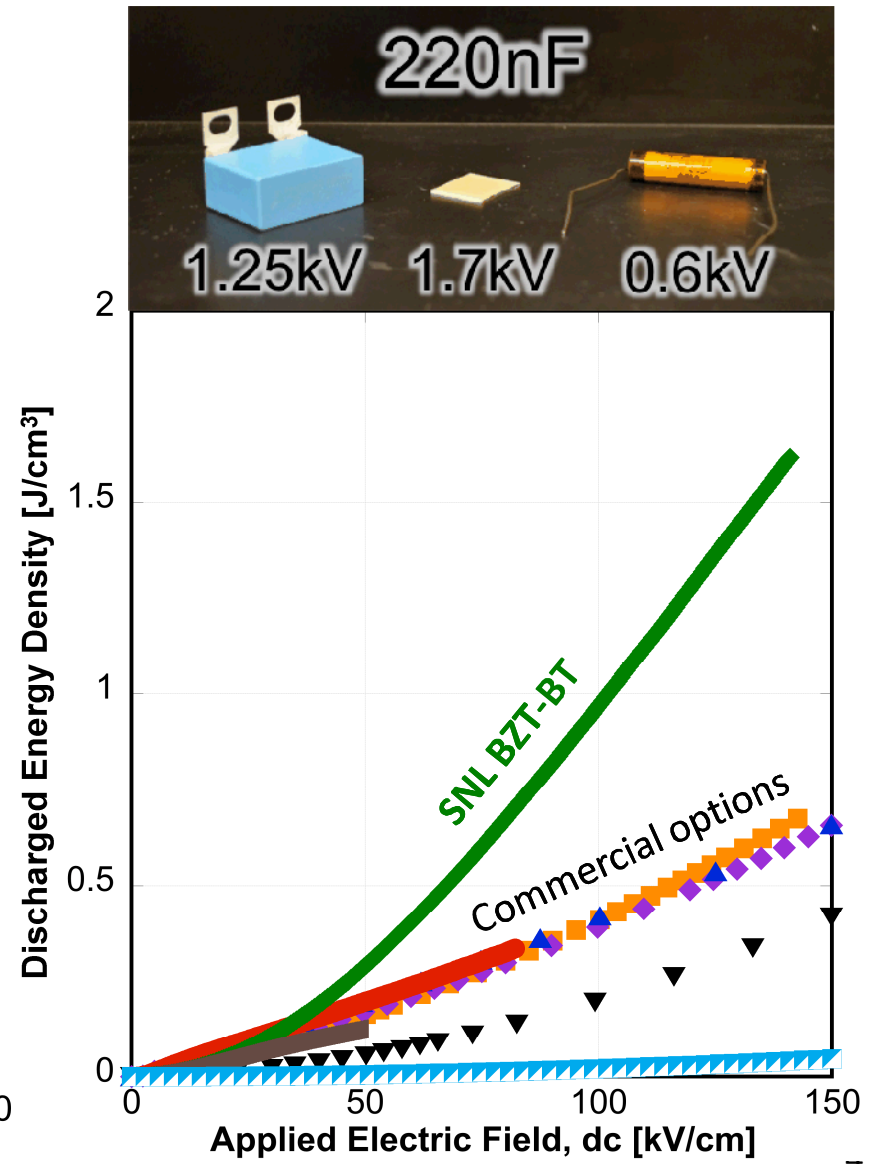
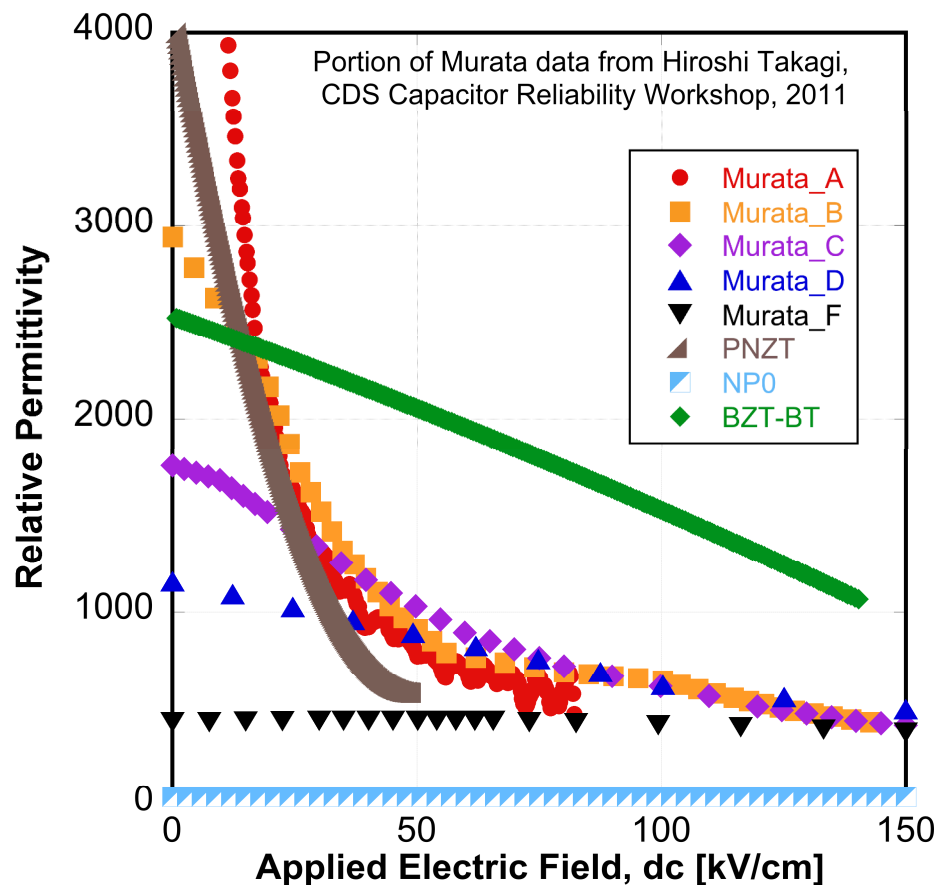
Pulse discharge applications require high energy density, low electromechanical response, low ESR



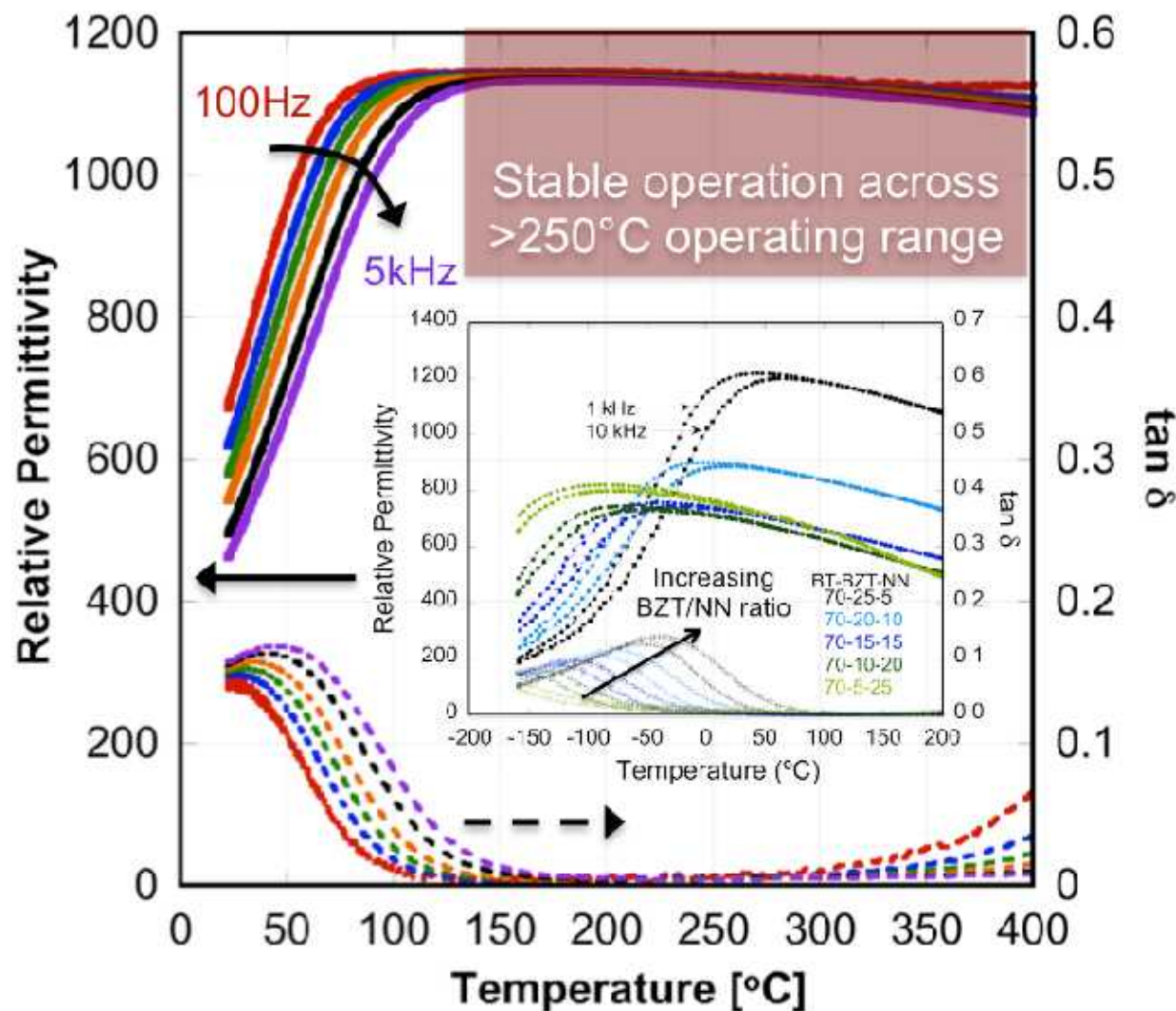
Simulation courtesy of
Eric Green and Stan Atcitty

High Energy Density Dielectrics

$$J = \int_0^{V_{\max}} CV^2 \rightarrow \int_0^{E_{\max}} KE^2$$



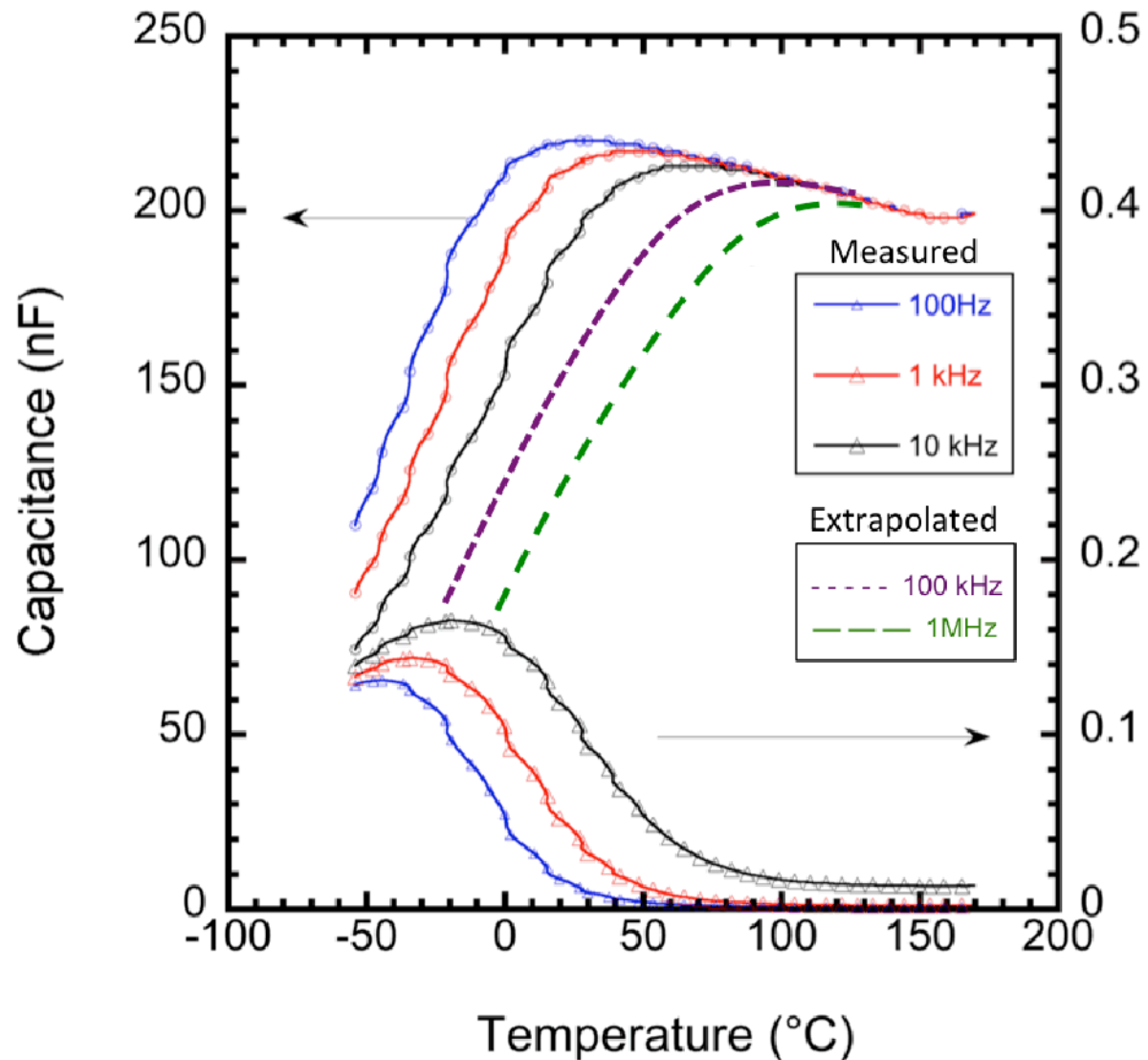
High Temperature Operation



- BiScO₃ stabilizes high temperature permittivity
- SrTiO₃ or NaNbO₃ additions shift relaxor transition to lower temperatures

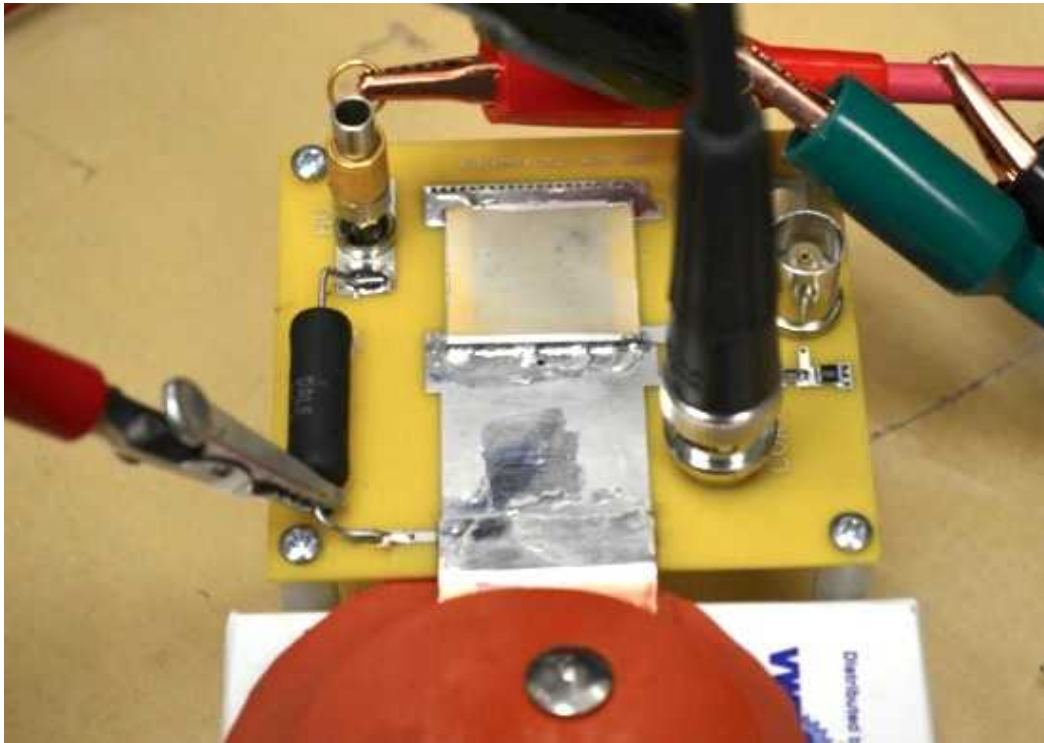
We can shift the temperature range for stable operation around by ~250°C via chemical modification *without sacrificing voltage stability*

Time Domain Performance



- Relaxor dielectrics exhibit characteristic frequency dispersion over relatively broad temperature ranges
- For pulse discharge or switched inverter applications that charge slowly and discharge quickly, which values are relevant?

Time Domain Performance

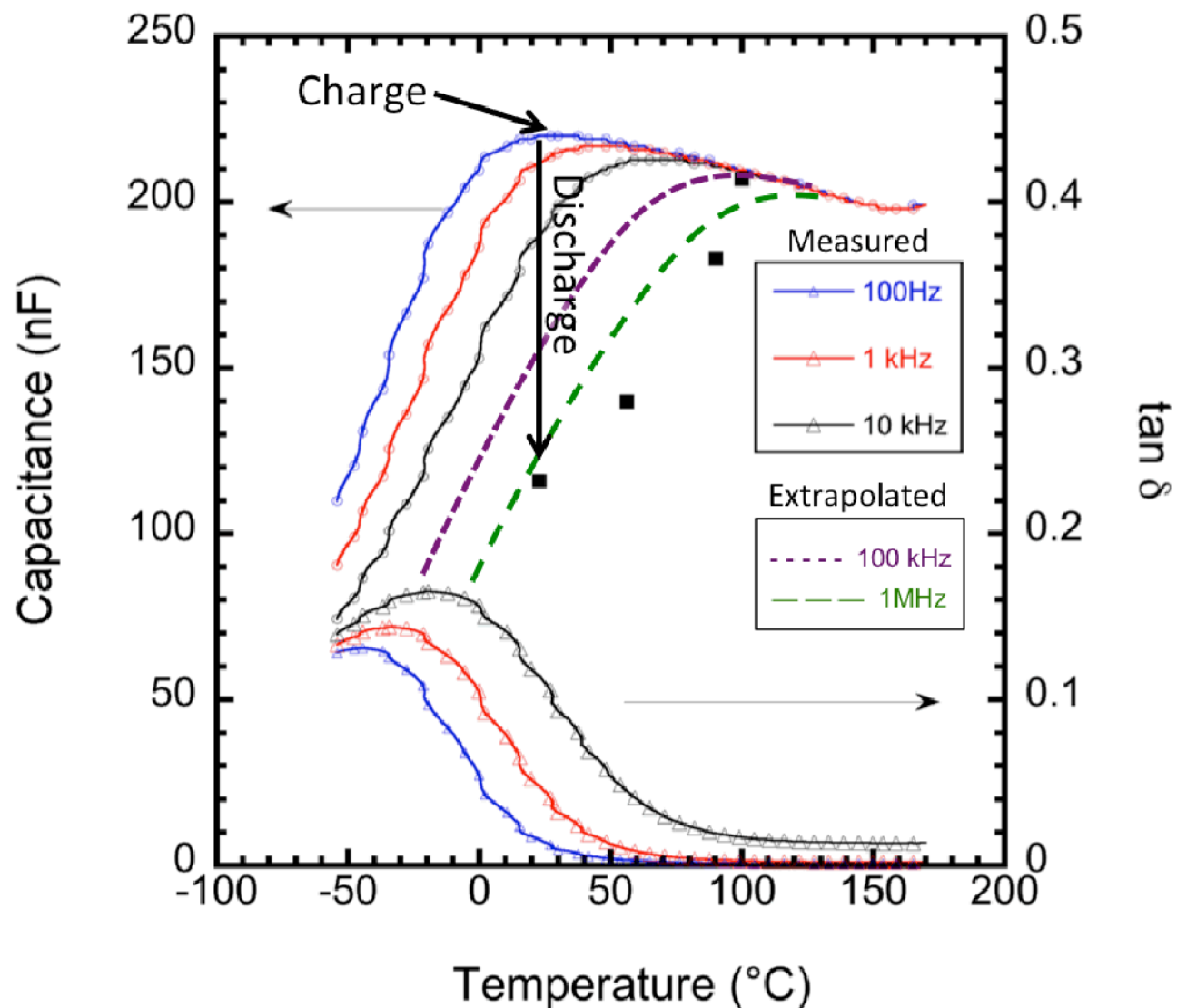


Low inductance board with compensated and calibrated CVR for pulse discharge (time domain) measurements

Large piezoelectric response of traditional relaxor ferroelectric causes fracture during such testing

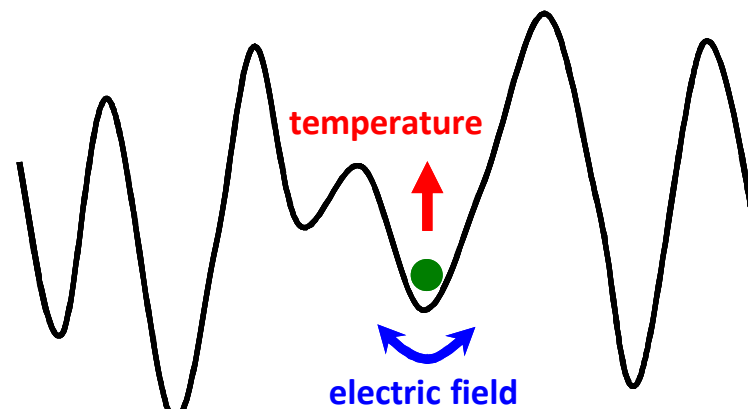
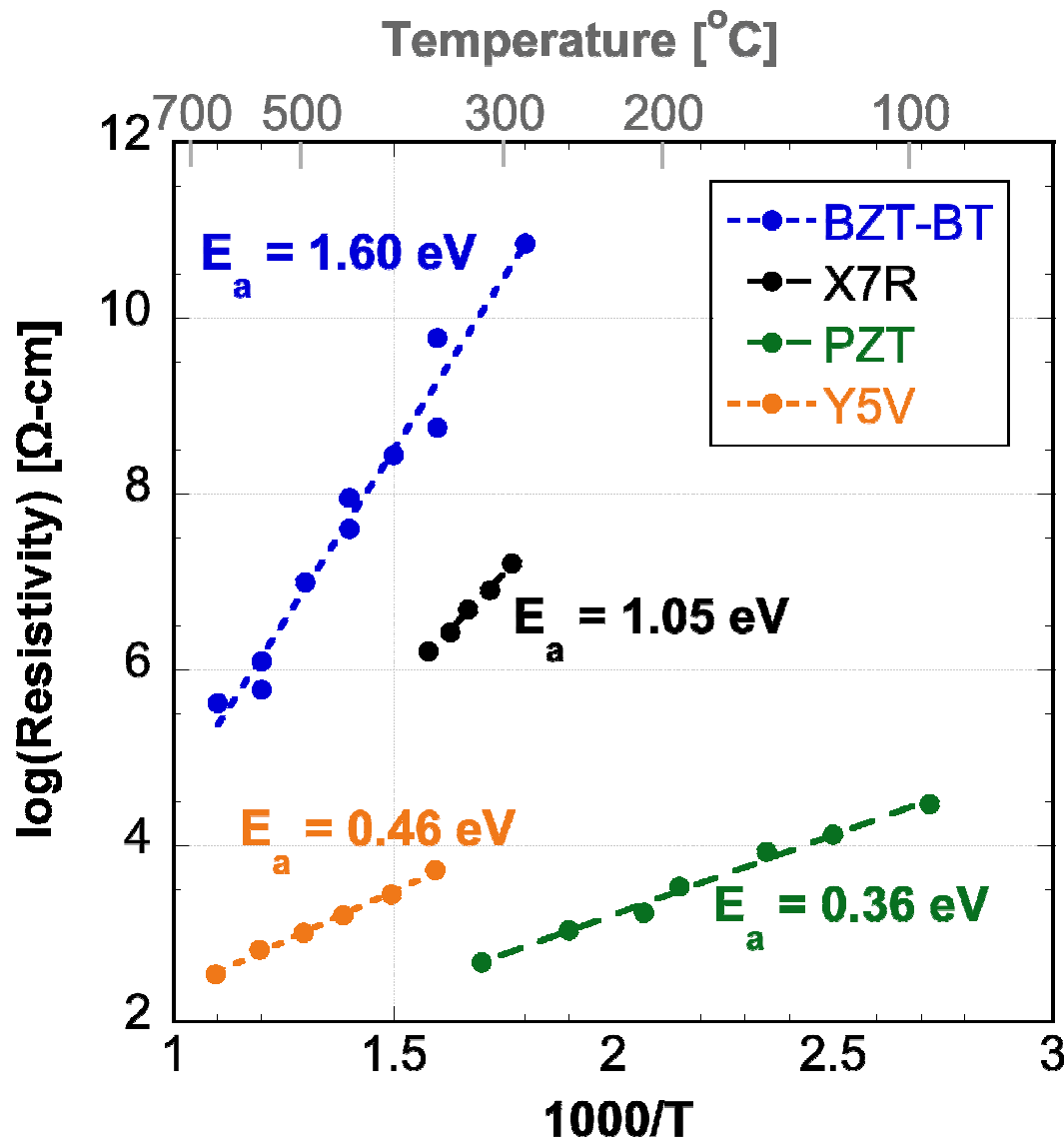
- Fitting critically damped discharge waveform allows determination of real part of permittivity

Time Domain Performance



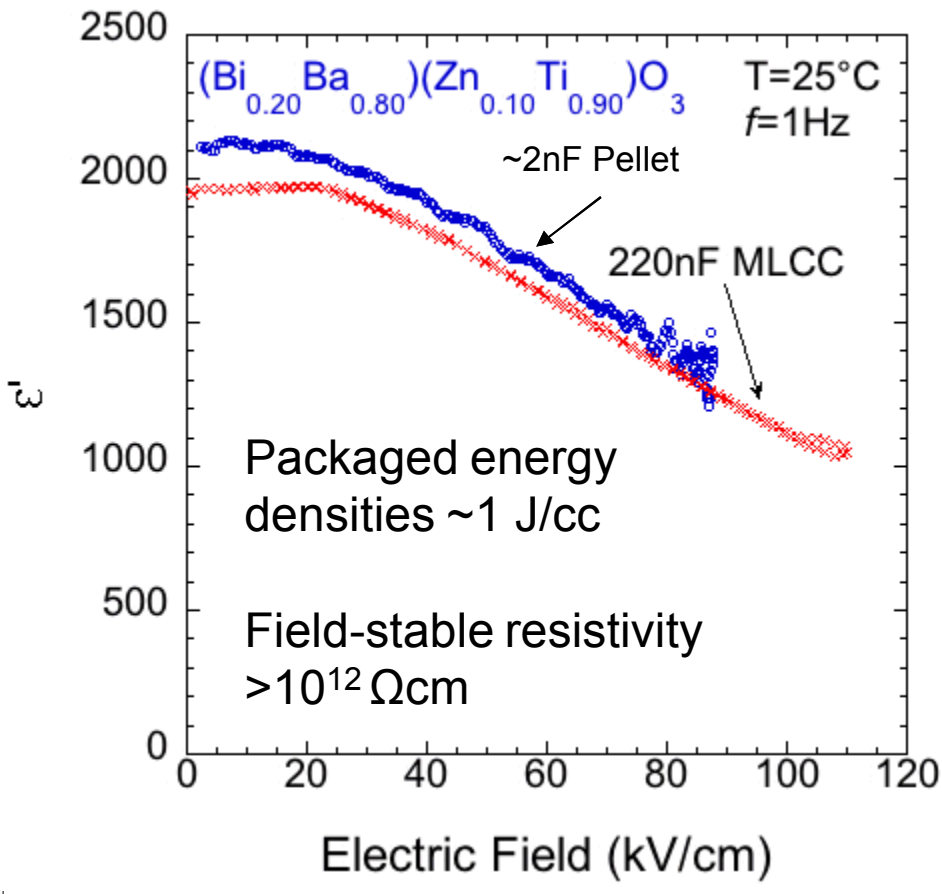
- Discharge data correspond to frequencies from 2.0-2.8 MHz
- Direct time domain results correlate well with extrapolated frequency-domain curves
- First known confirmation of relaxor response in time domain

High dc Resistivity → Reliable



Higher resistivity and larger activation energy for conduction both translate into longer lifetimes and higher reliability, particularly at elevated temperatures.

Prototype MLCCs: 200nF @ 1700V

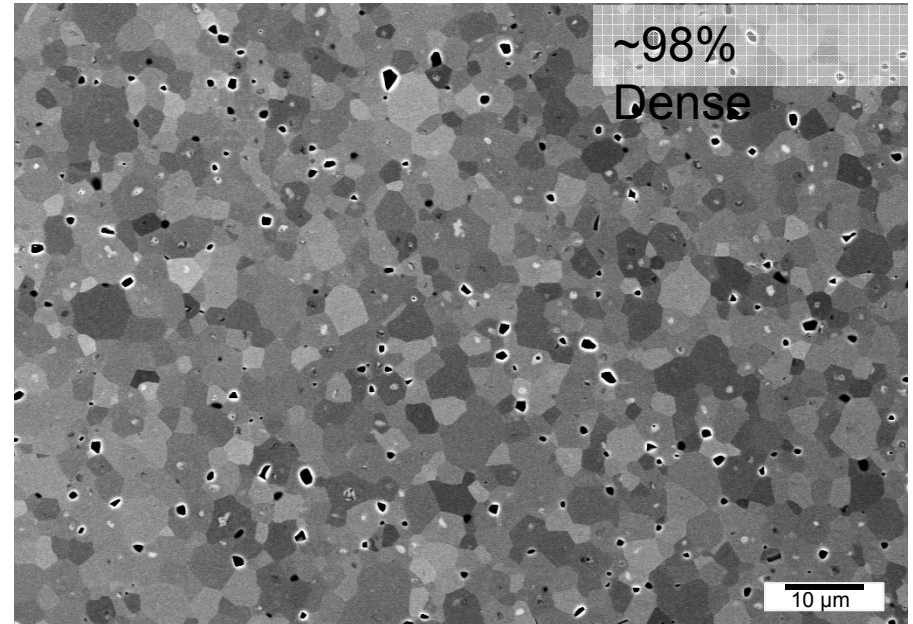
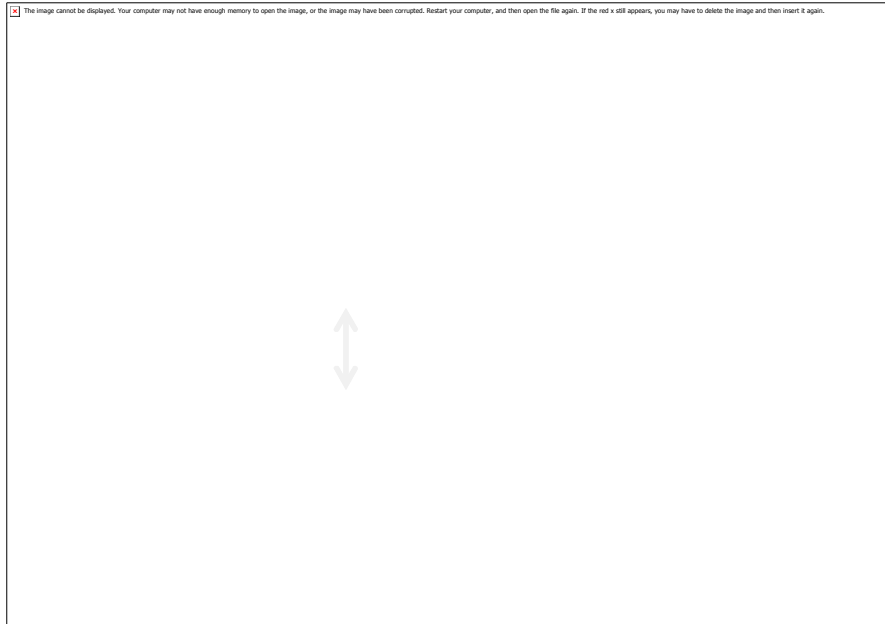


Cast and co-
fire with Pt
@ 1040°C

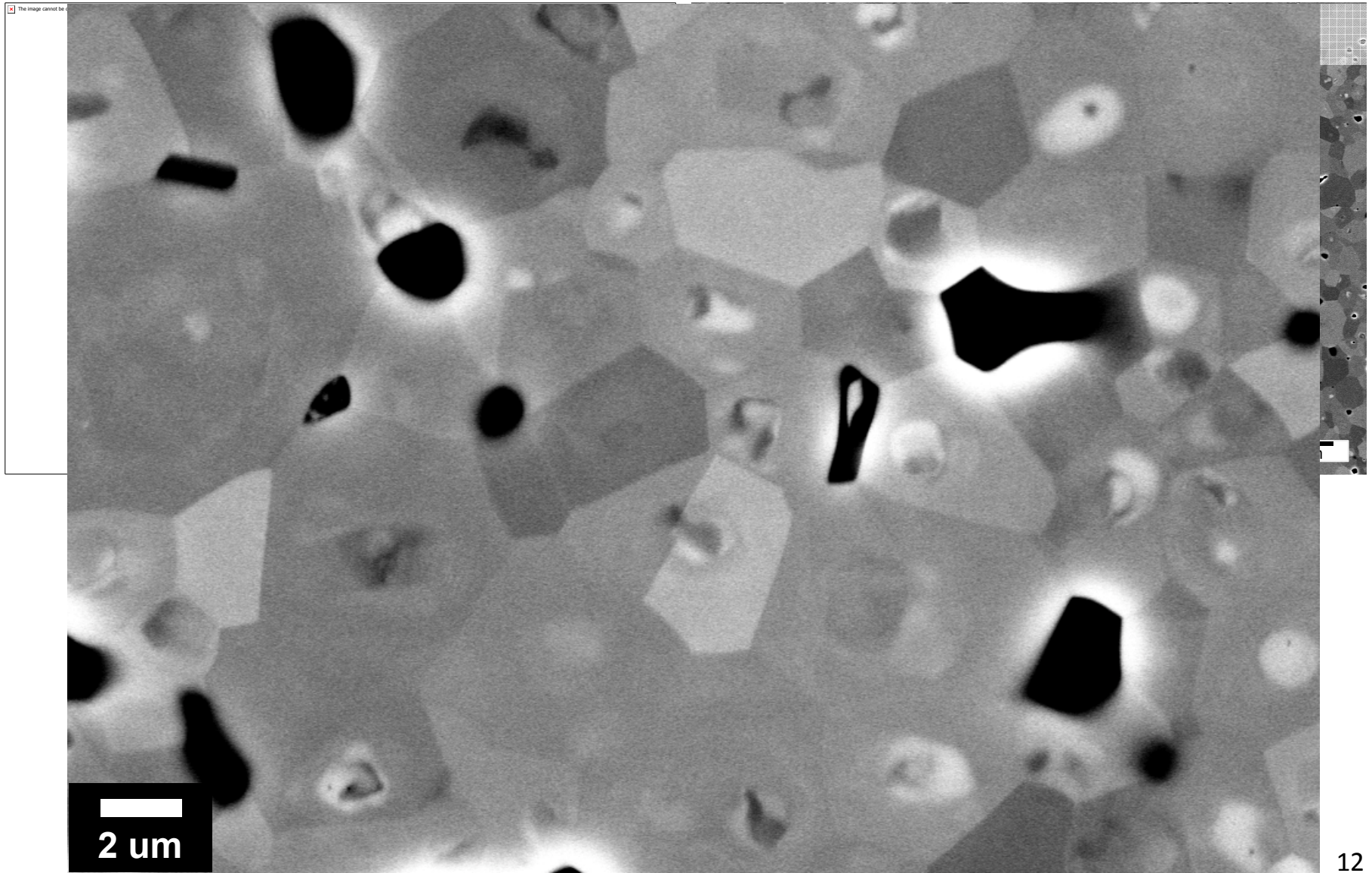


- Maintain RC time constants $> 200 \text{ sec}$ even at $> 60 \text{ kV/cm}$, $> 150^\circ\text{C}$ (likely higher)
- Smaller samples tested under harsher conditions maintain RC $> 100 \text{ sec}$ $> 350^\circ\text{C}$
- Mechanism(s)??

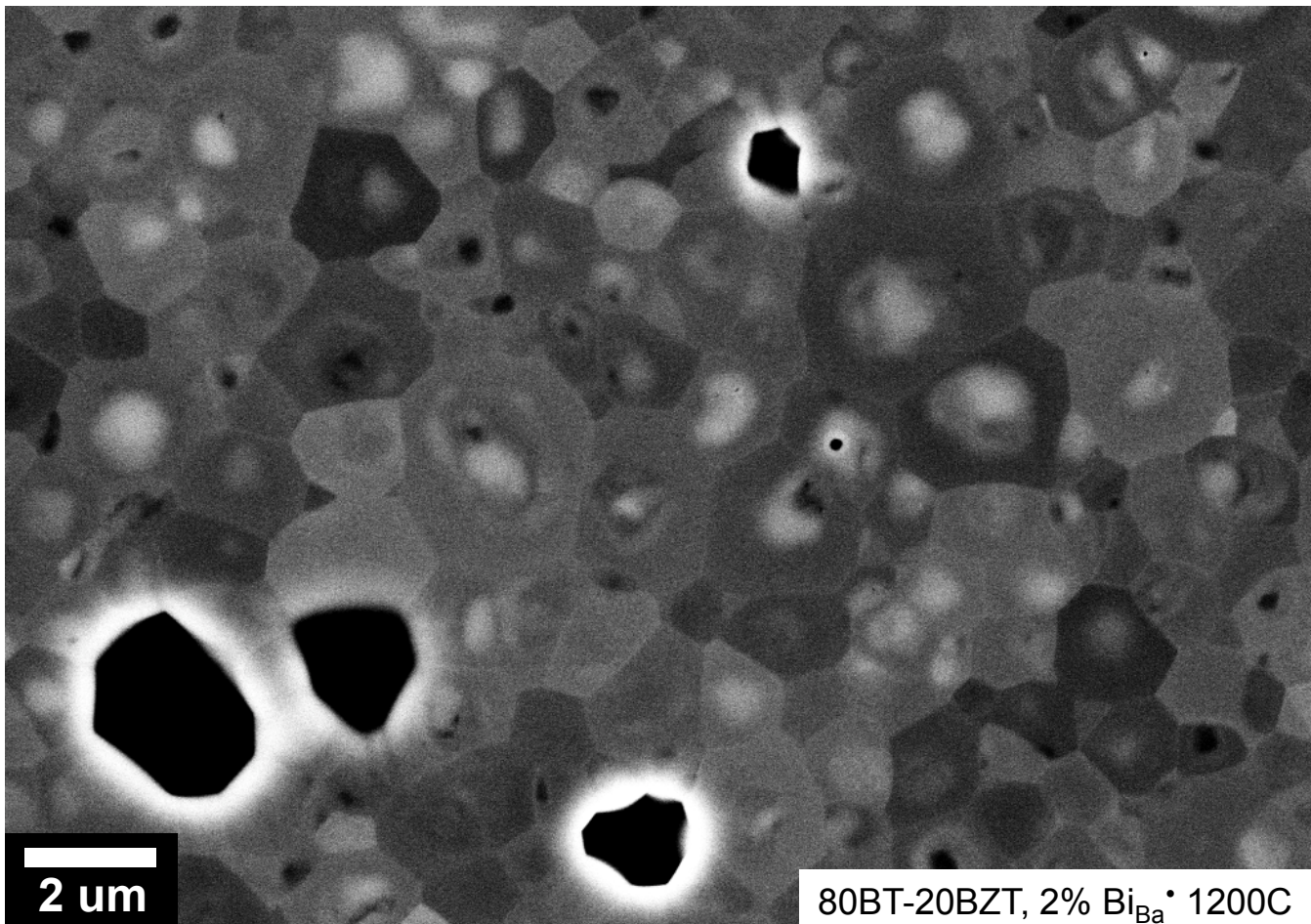
MLCC dielectric exhibits high-Z core



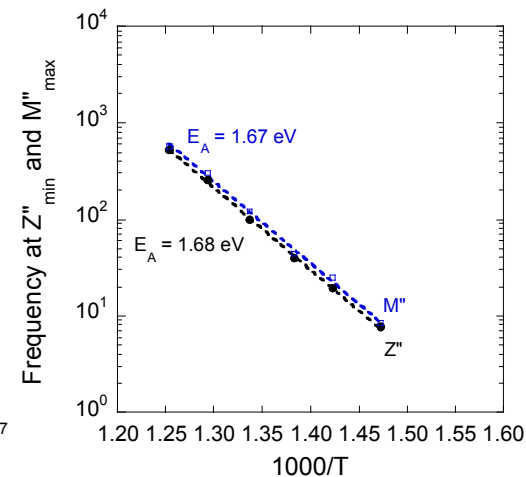
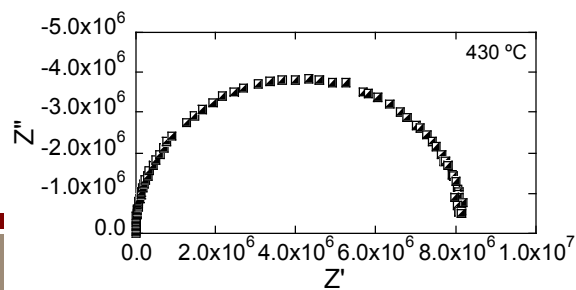
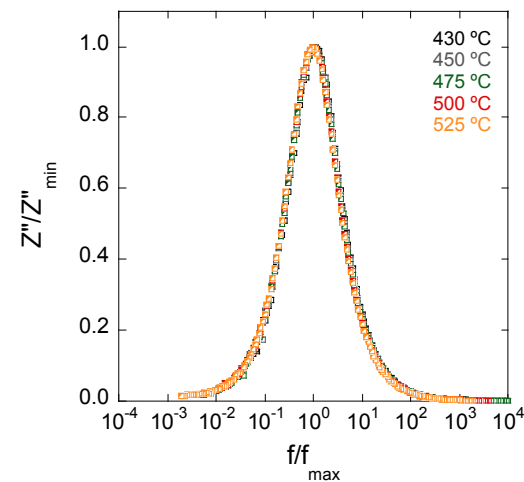
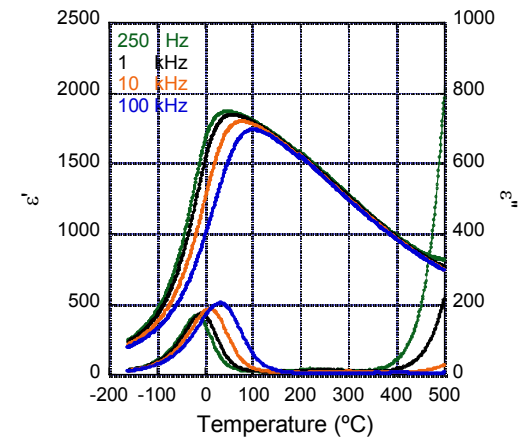
MLCC dielectric exhibits high-Z core



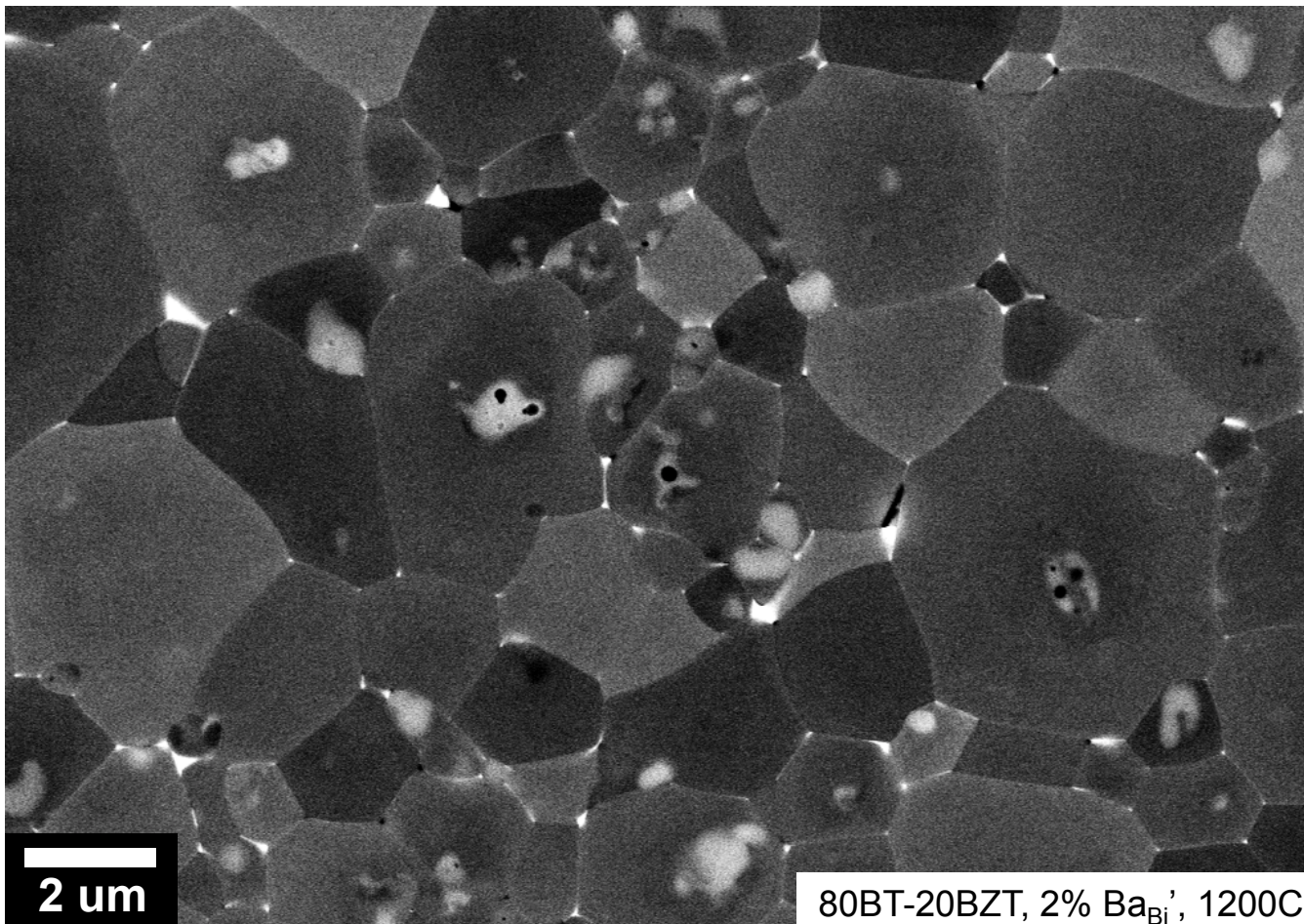
Donor-doped



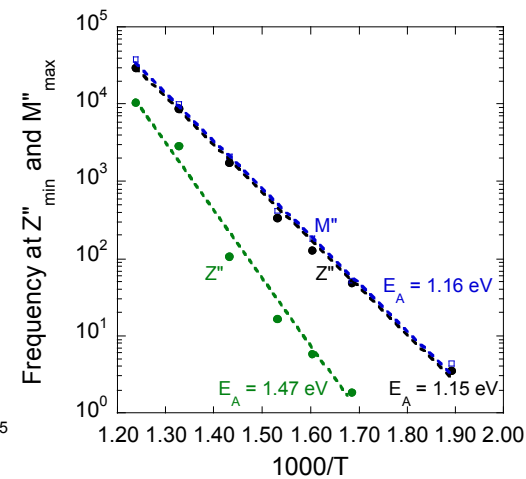
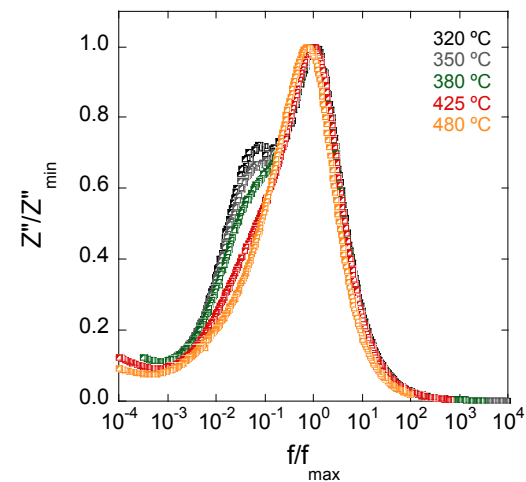
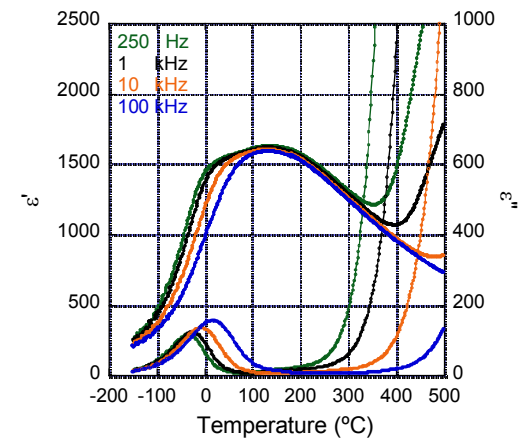
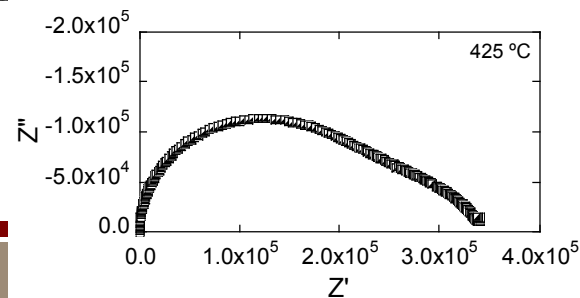
80BT-20BZT, 2% Bi_{Ba} • 1200C



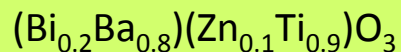
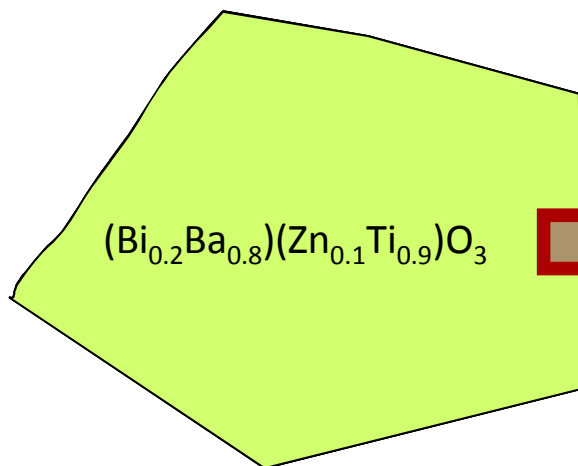
Acceptor-doped



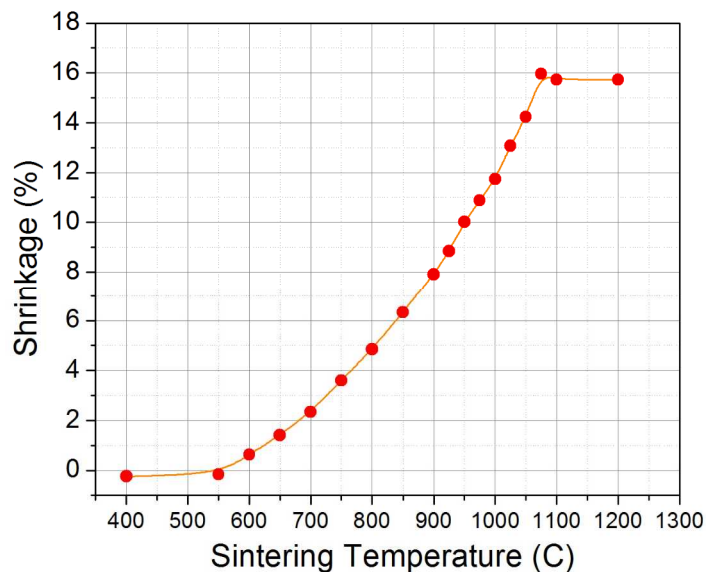
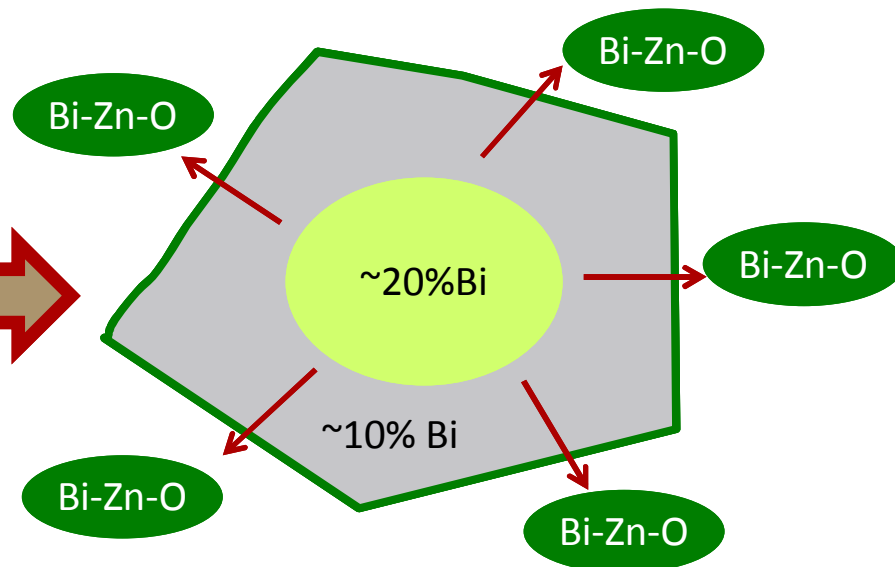
80BT-20BZT, 2% BaBi', 1200C



Calcination



At Calcination temp (950 C)

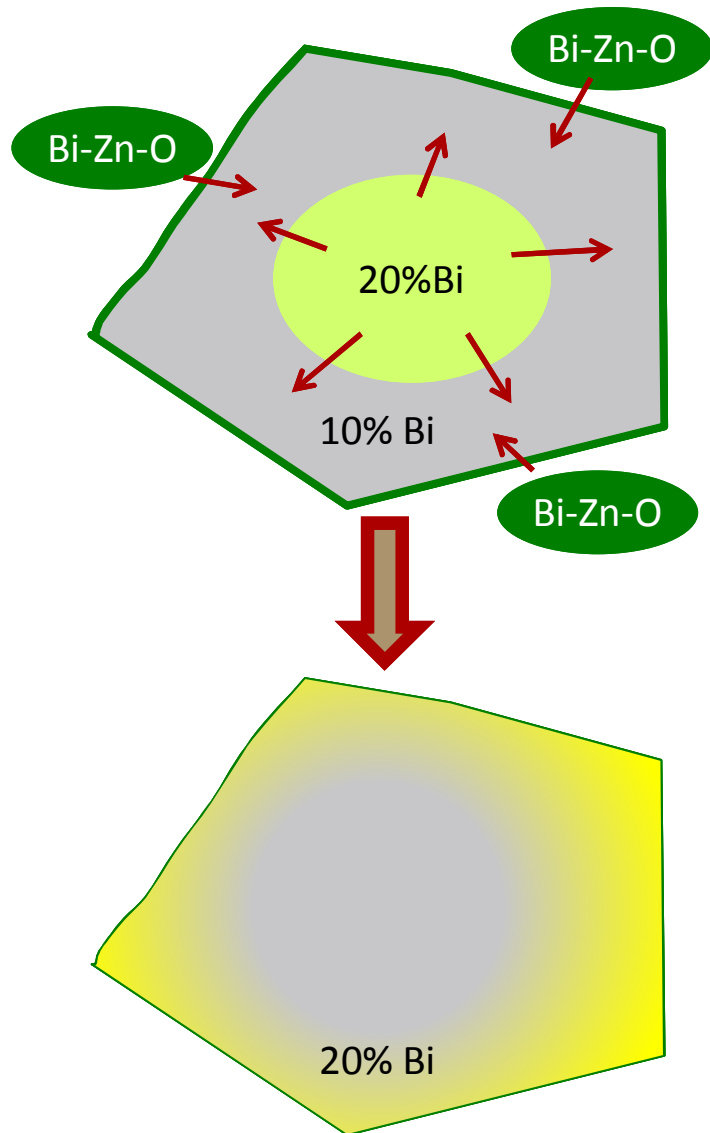


Working assumption is that the equilibrium solubility of Bi_2O_3 in BaTiO_3 is breached upon cooling.

On cooling from 950 C, Bi exsolves from perovskite solid solution, leading to Bi-rich core and Bi-Zn-O glassy phase at particle surfaces.

Initial sintering occurs as Bi-Zn-O phase melts.
Next step depends on dominant defect species.

Sintering: Donor doped



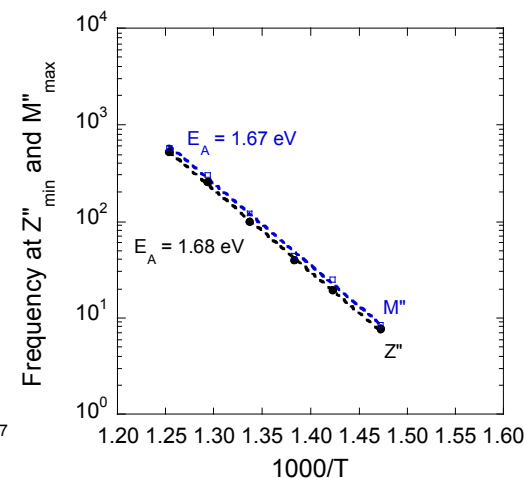
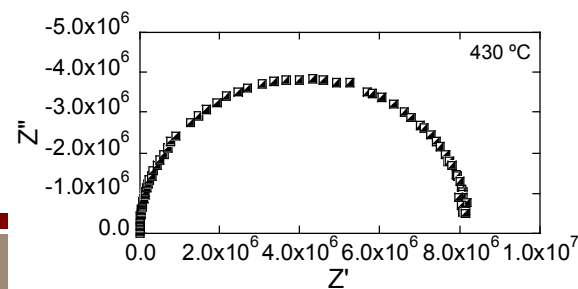
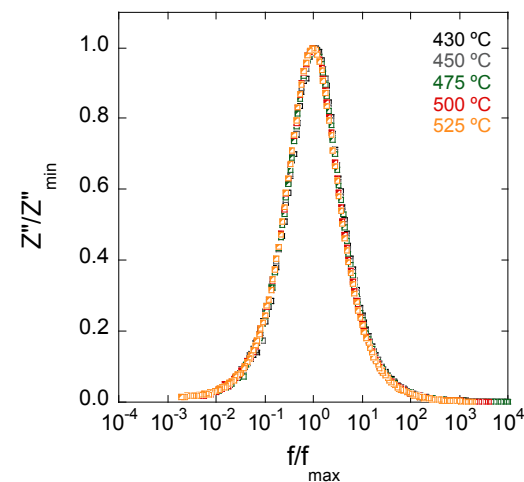
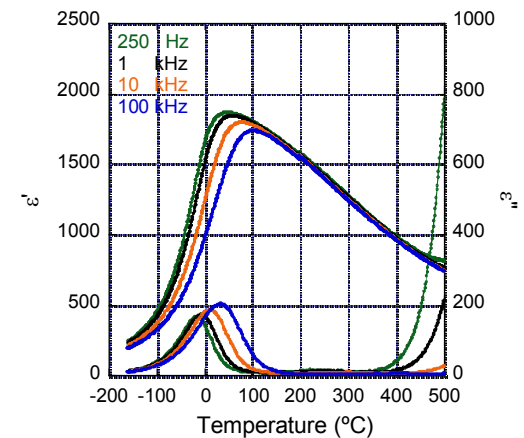
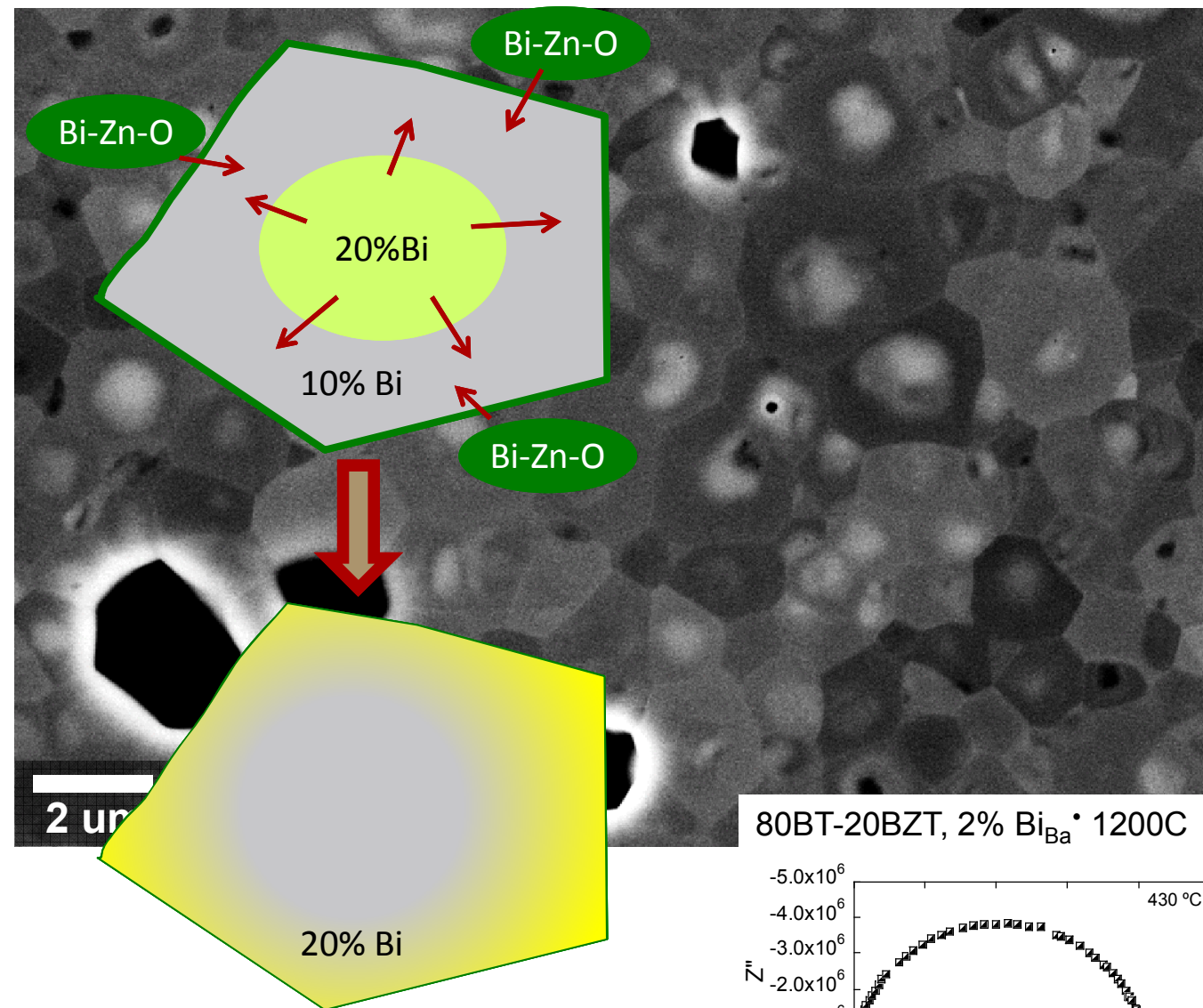
Donor Doped: Cation Vacancies Dominate

- Bi diffuses in from g.b., out from core
- Diffusion **assisted** by cation vacancies

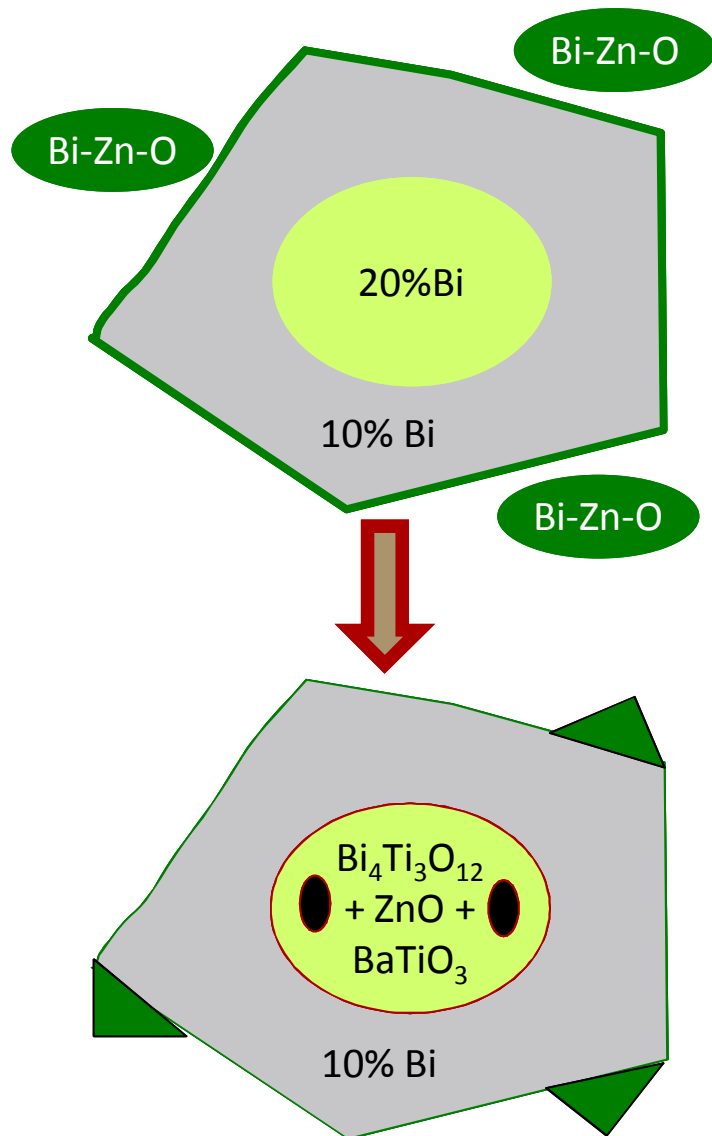
Resultant Microstructure

- Relatively homogenous
- Bi-gradient with diffuse boundary due to ample Bi diffusion
- Electrically homogeneous microstructure (single relaxation)

Sintering: Donor doped



Sintering: Acceptor doped



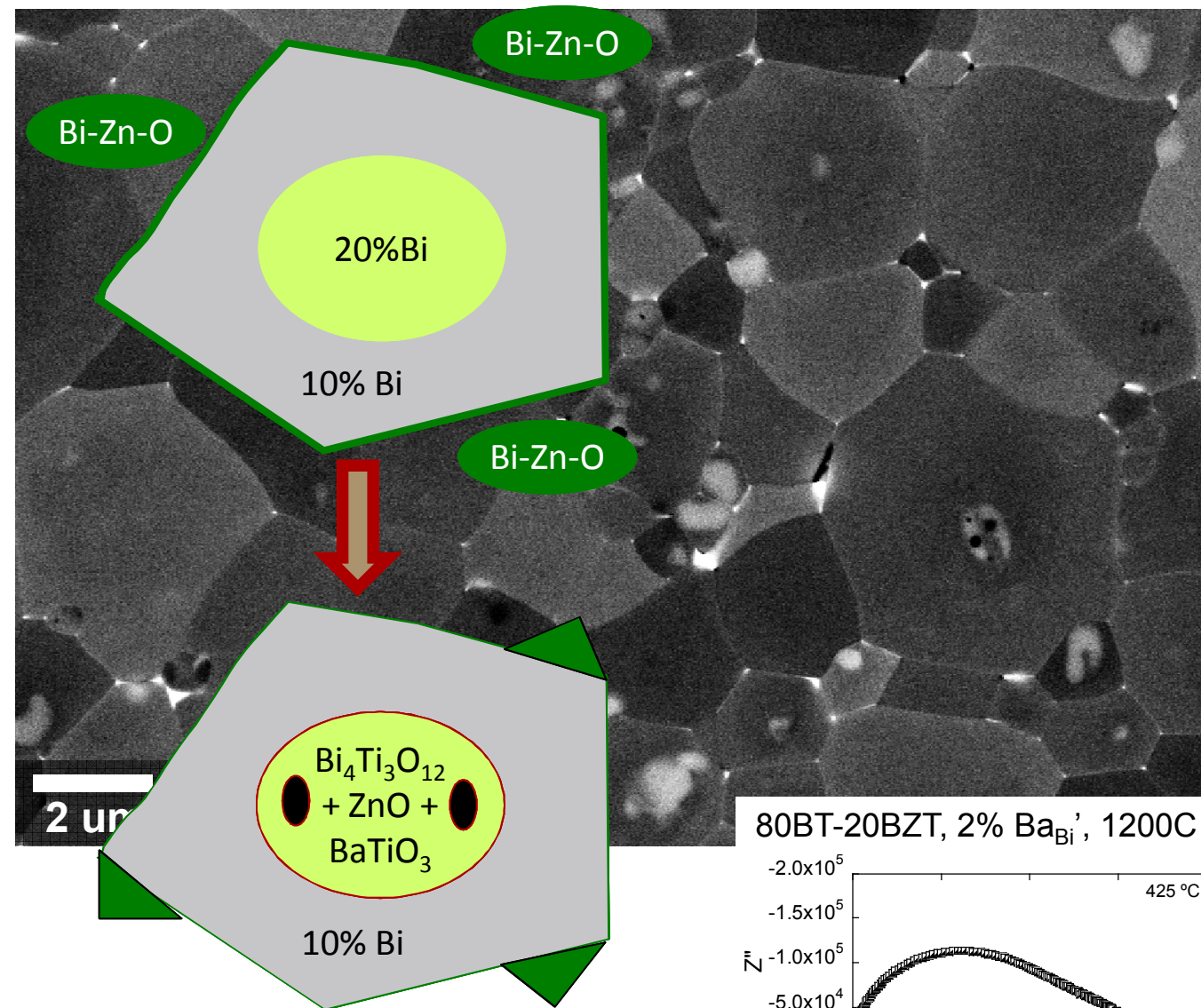
Acceptor Doped: Oxygen Vacancies Dominate

- Bi diffuses in from g.b., out from core
- Diffusion **inhibited** by lack of cation vacancies
- Bi-rich phase at triple points
- Bi-rich cores decompose into equilibrium phases: $\text{Bi}_4\text{Ti}_3\text{O}_{12} + \text{ZnO} + \text{Ba-Ti-O}$

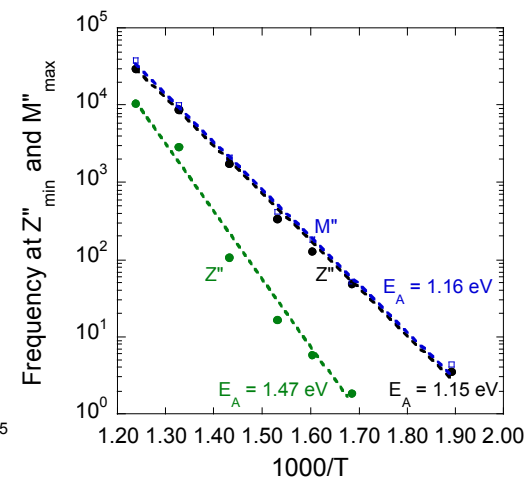
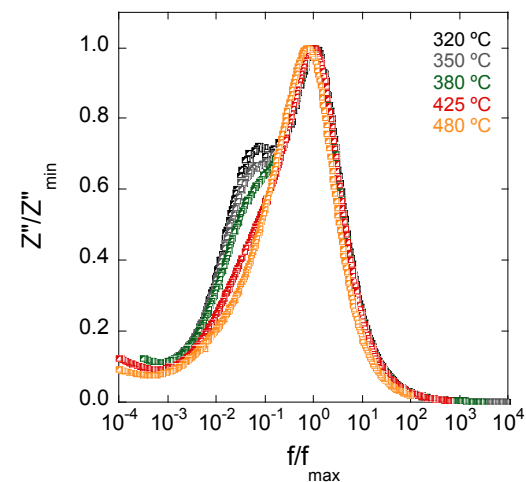
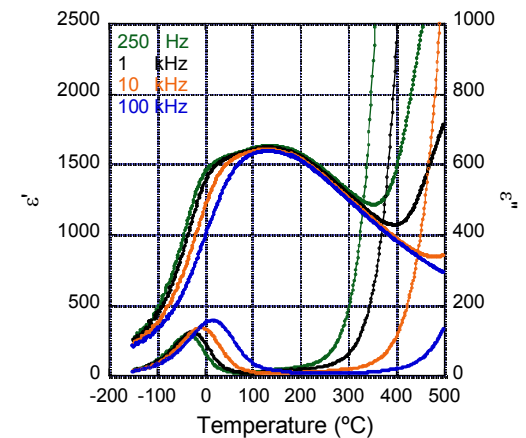
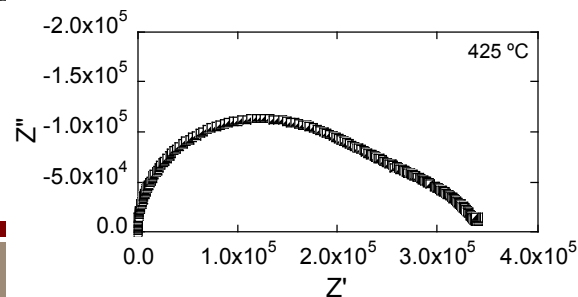
Resultant Microstructure

- High-Z phase at triple points
- Well defined core boundary
- Low-Z precipitates in core region
- Electrically heterogeneous microstructure (two relaxations)

Sintering: Acceptor doped

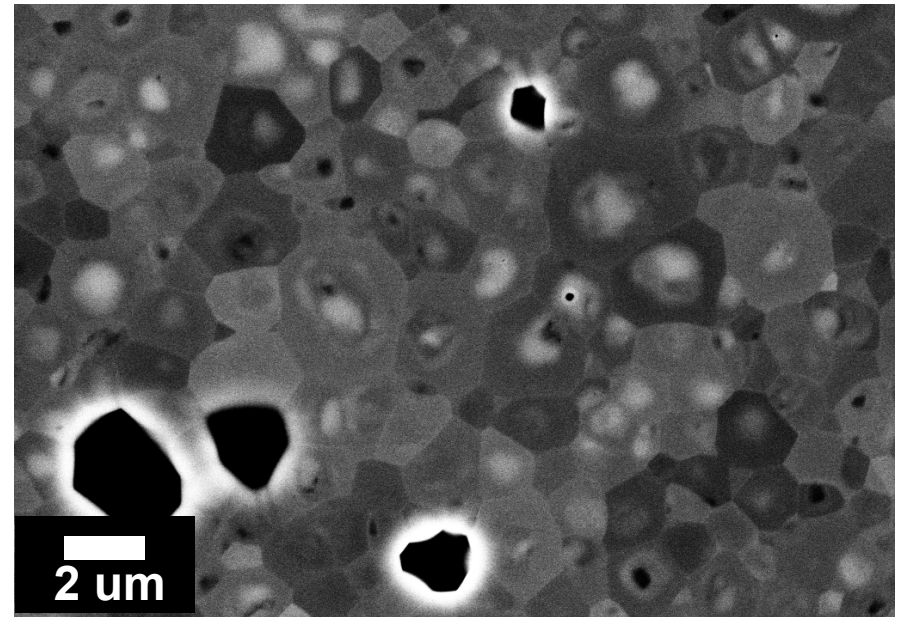
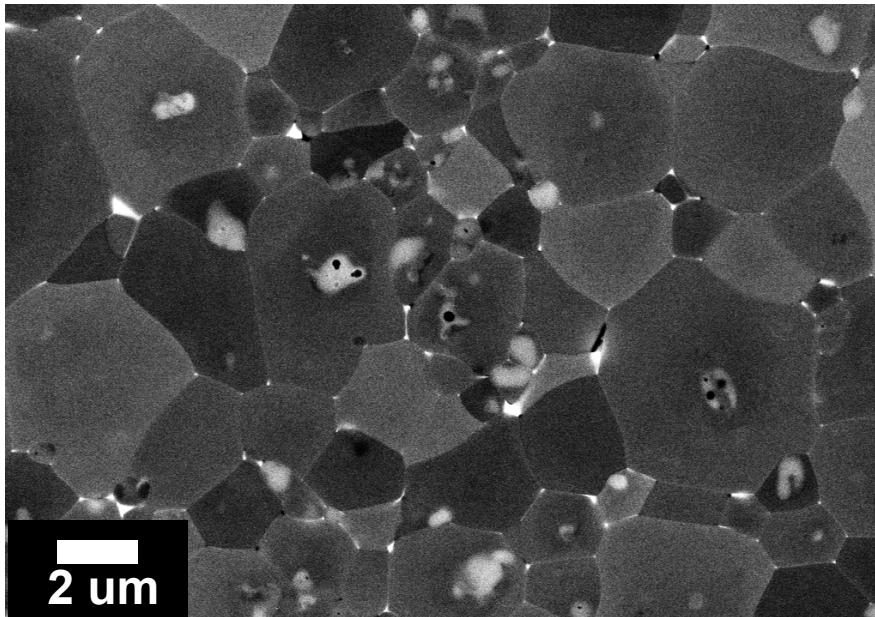
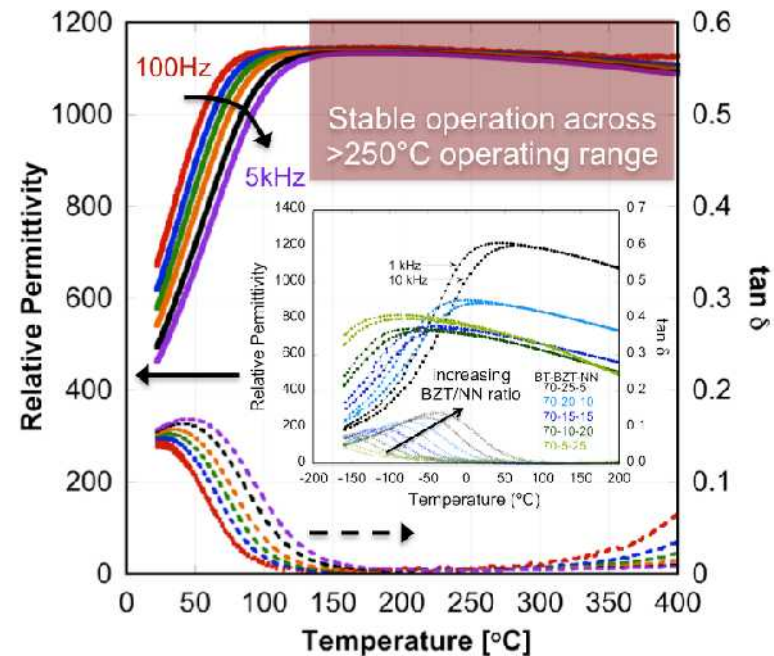


80BT-20BZT, 2% Ba_{Bi}' , 1200C

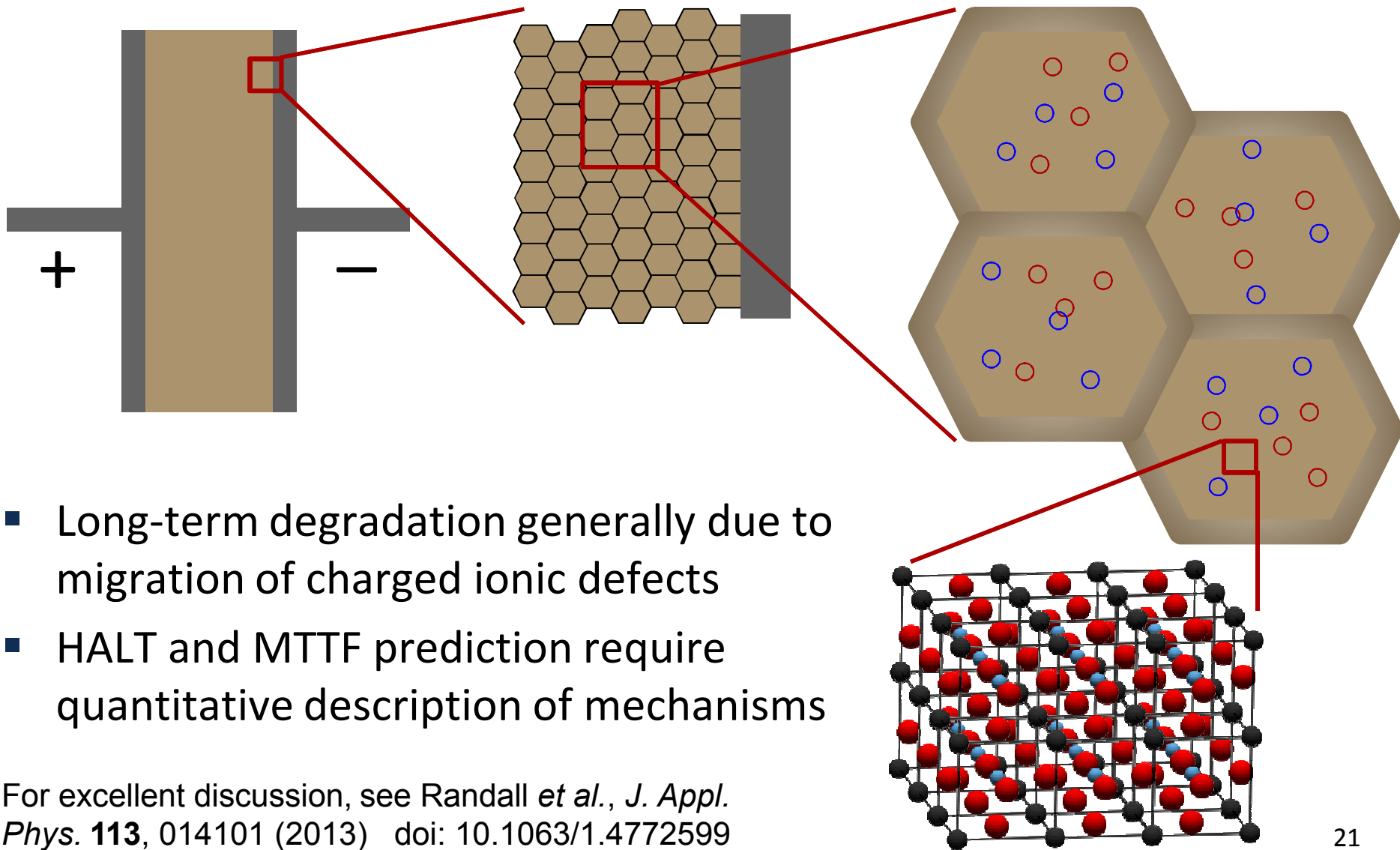


Summary

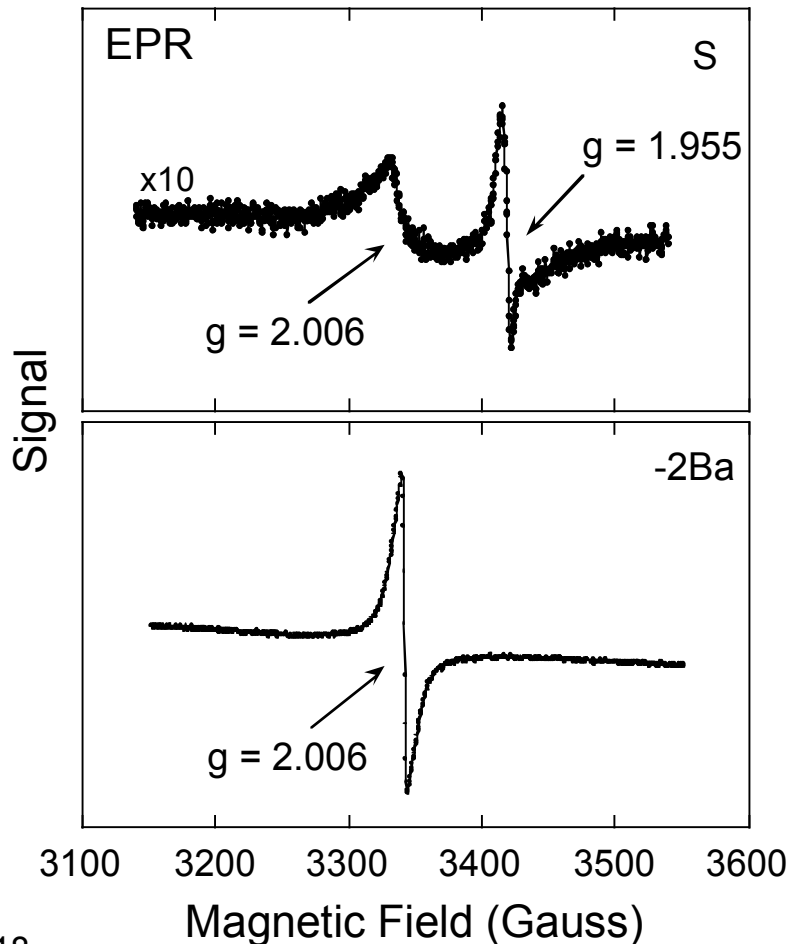
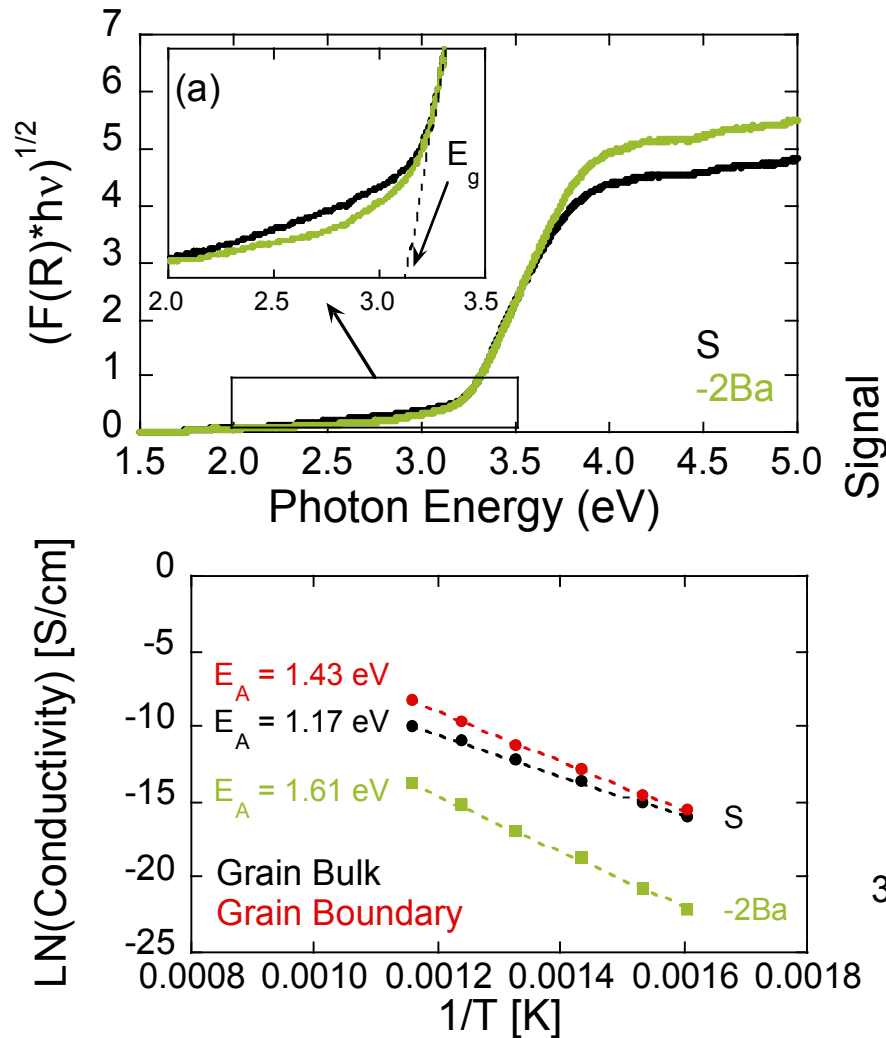
- Fabricated high energy density dielectrics with temperature- and voltage-stable performance
- Defect chemistry has strong influence on microstructure as well as electrical response



Degradation in Ceramic Dielectrics



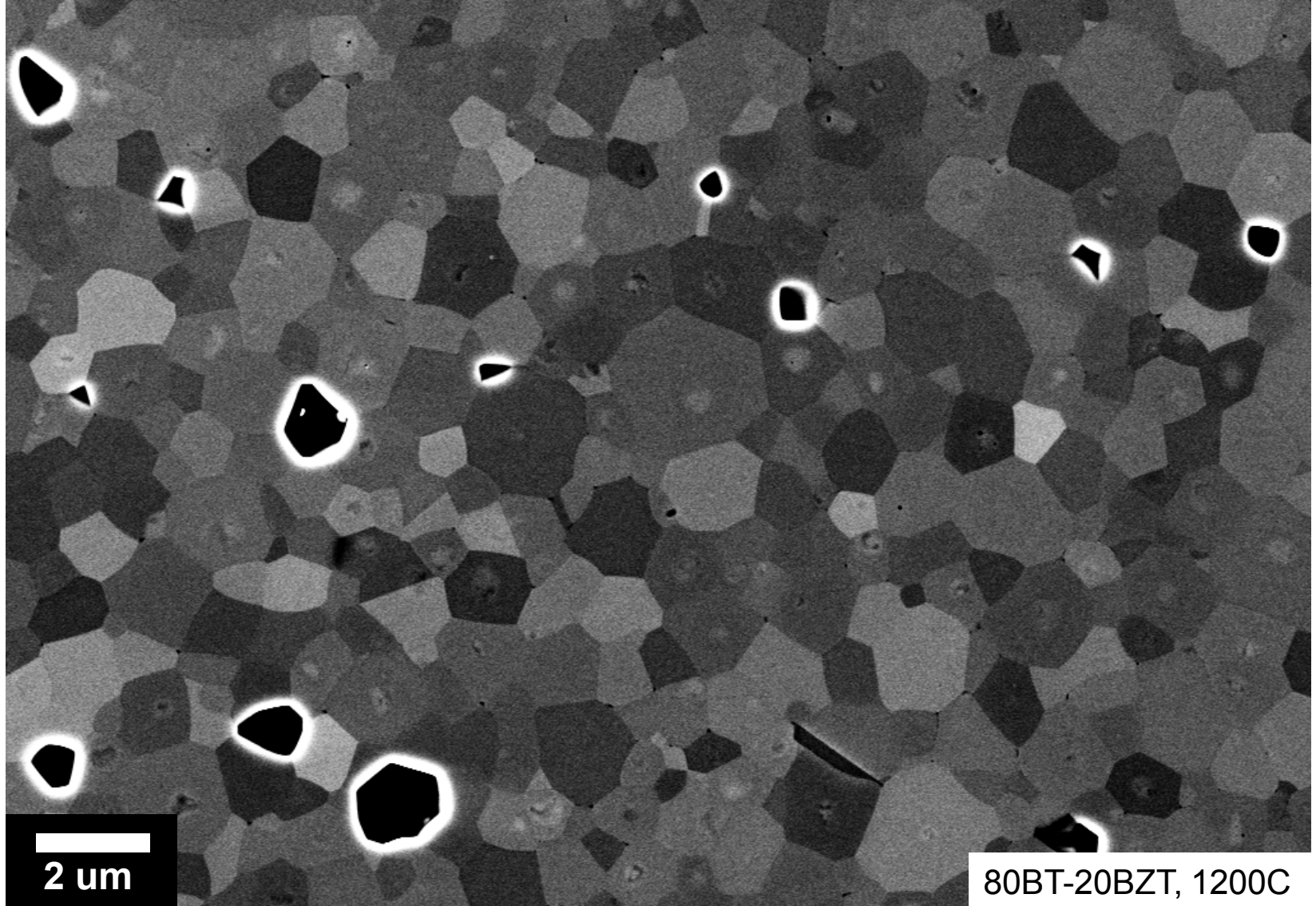
Defects: Optical & Electrical Info



Raengthon, et al., Appl. Phys. Lett. (2012)

- Defect studies suggest $V_{Ba}'' - V_O^{\bullet\bullet}$ pairs are strong carrier traps

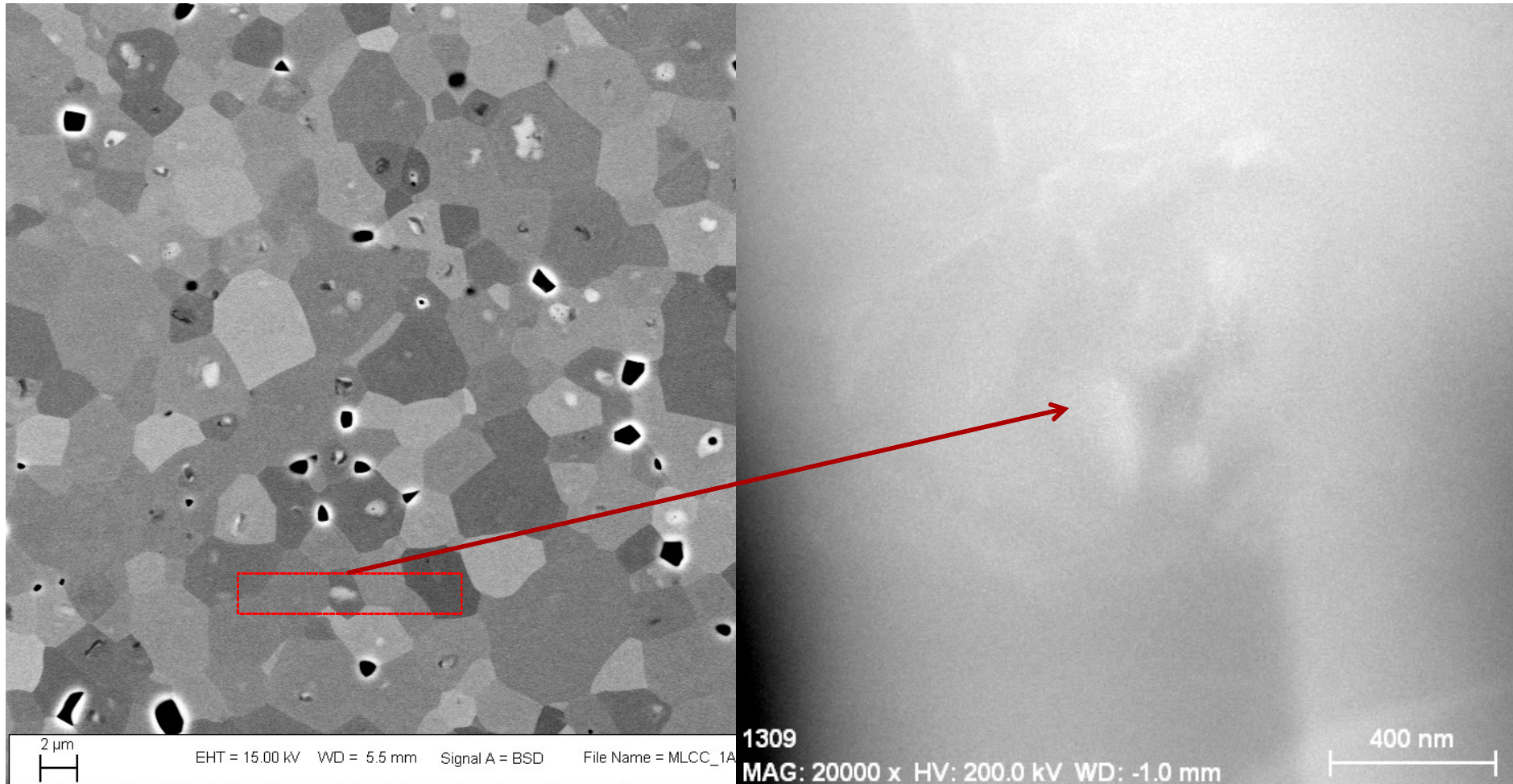
Well-Polished Section, Channeling



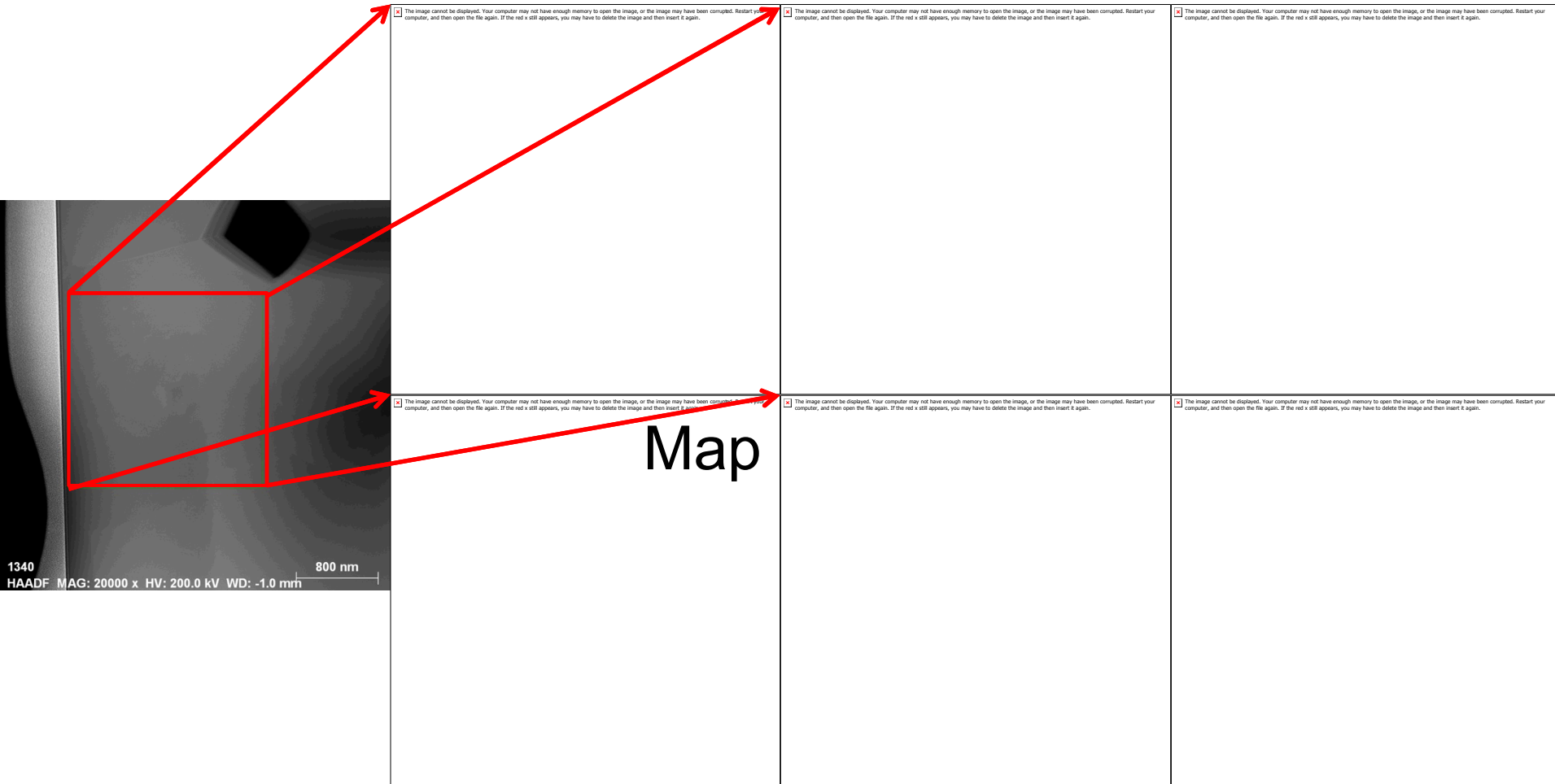
Compositional Variation

SEM

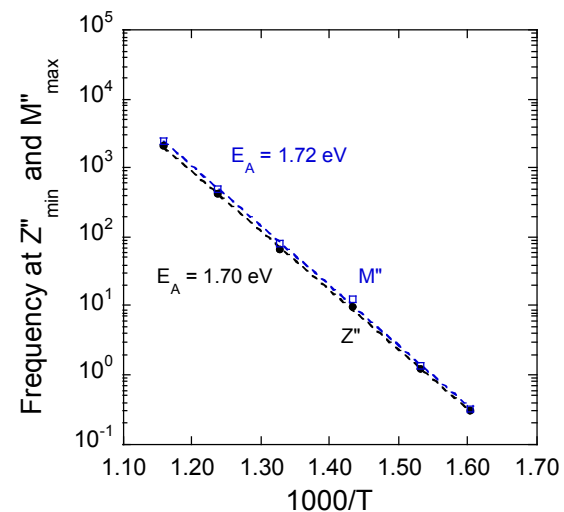
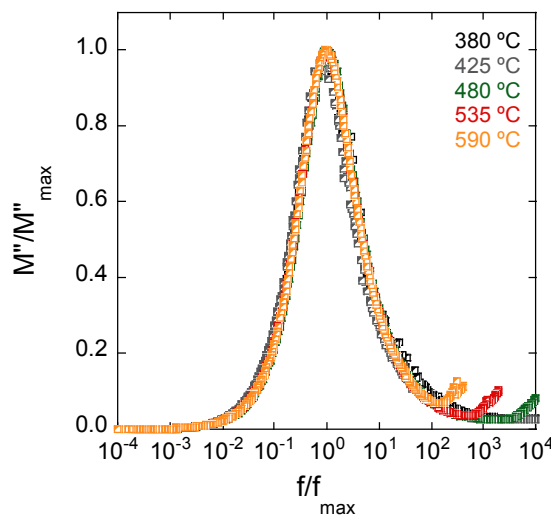
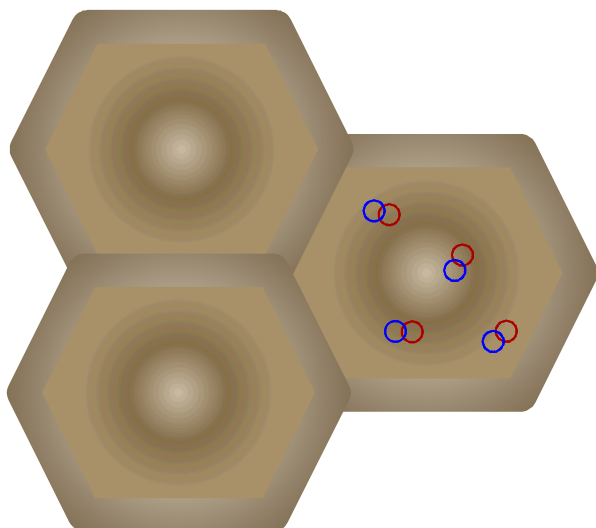
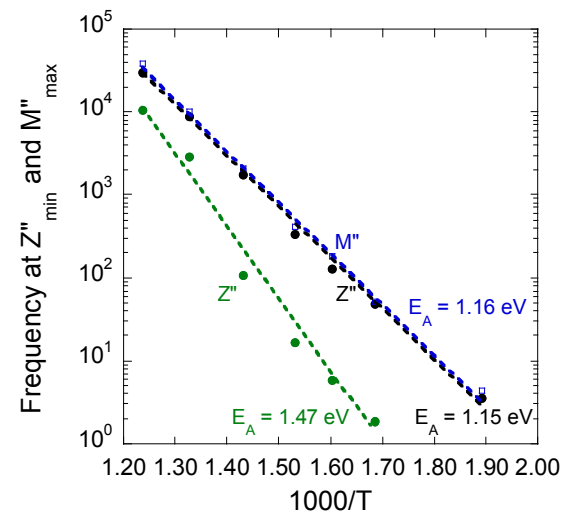
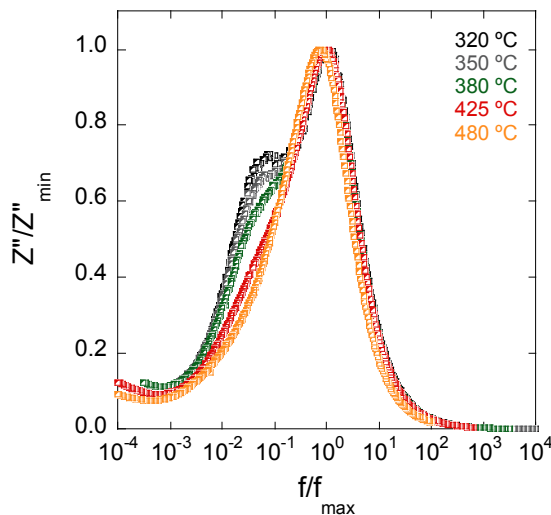
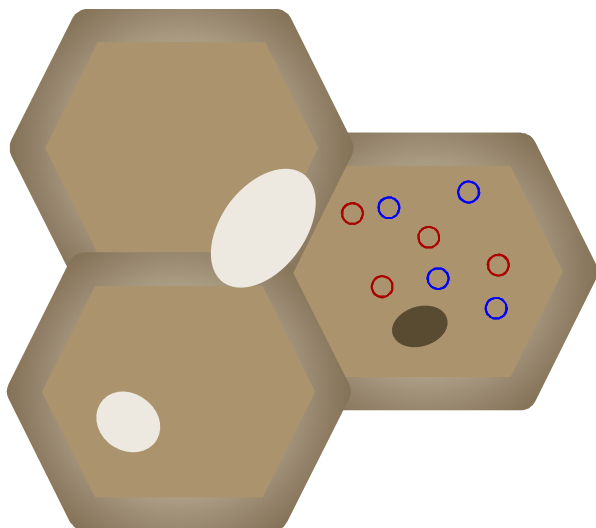
HAADF TEM



Bi and Zn Co-segregation



Microscale Heterogeneity



with Prof. David Cann, Oregon State University