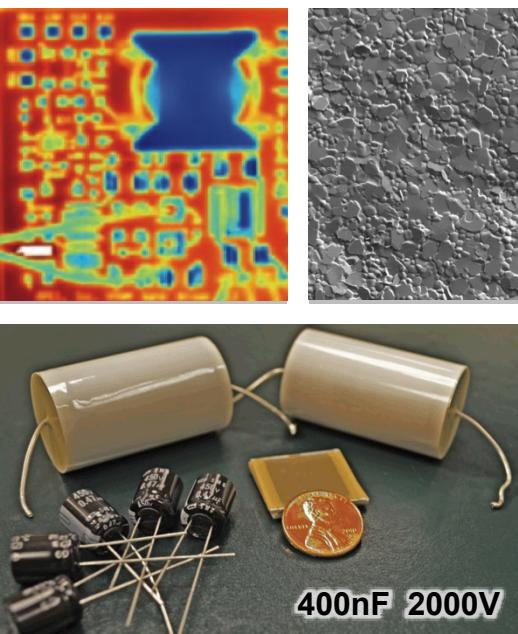


Ceramic Capacitors based on Temperature and Voltage-Stable Relaxor Dielectrics



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John Borchardt, and Mia Blea-Kirby

Sandia National Laboratories

Natthaphon Raengthon and David Cann

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Kevin Ring

New Mexico Tech



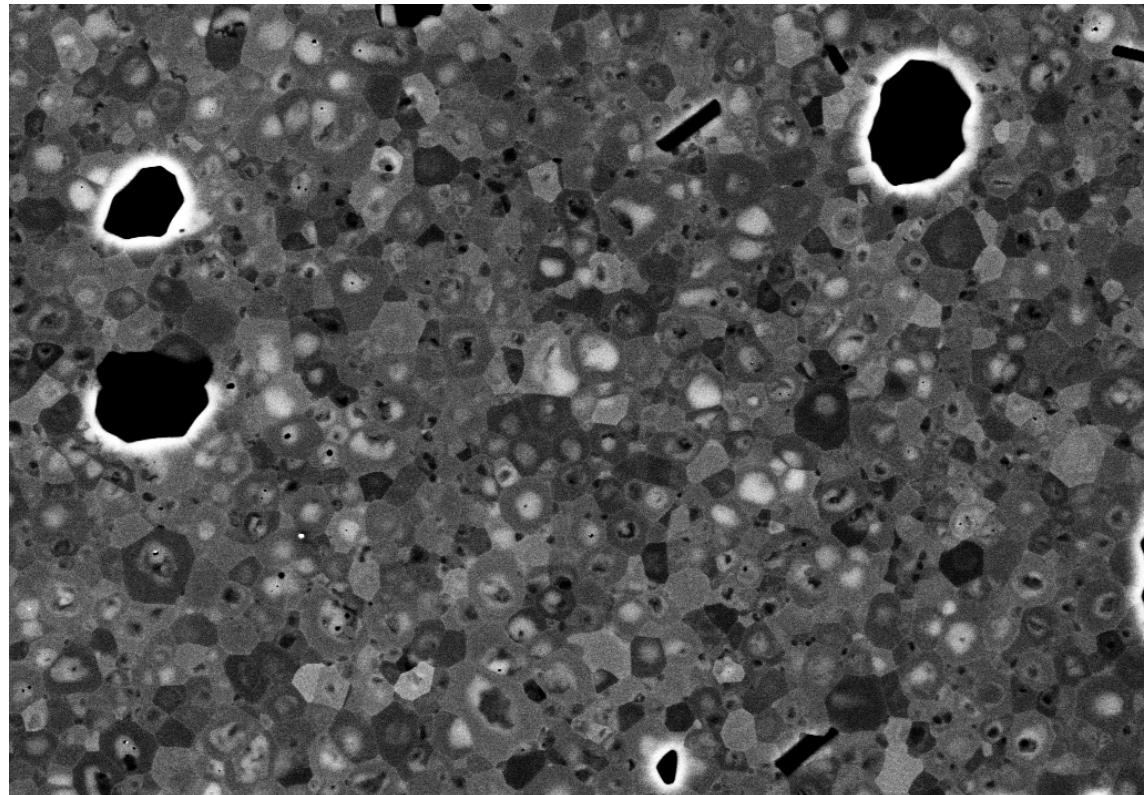
*Exceptional
service
in the
national
interest*



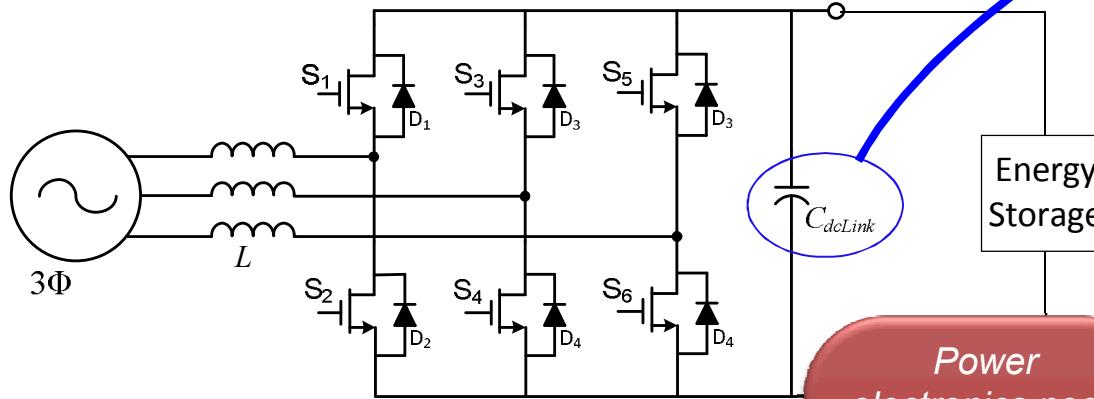
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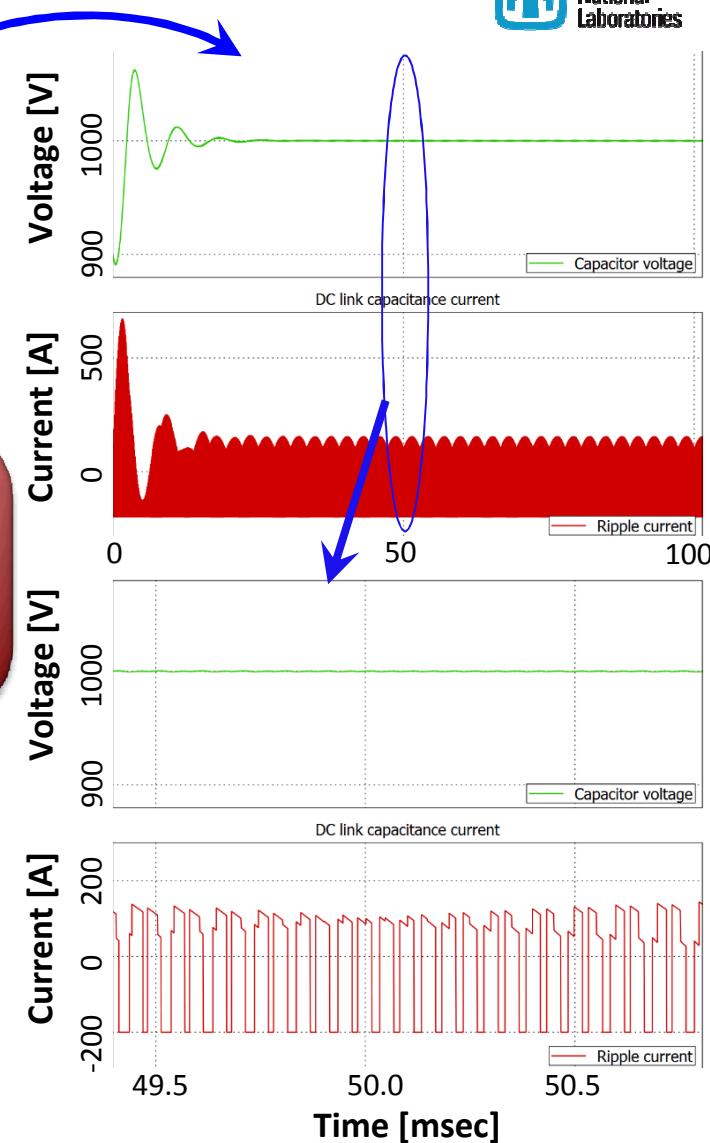


Application Space



Power electronics need high operating temperature, low ESR, high resistivity

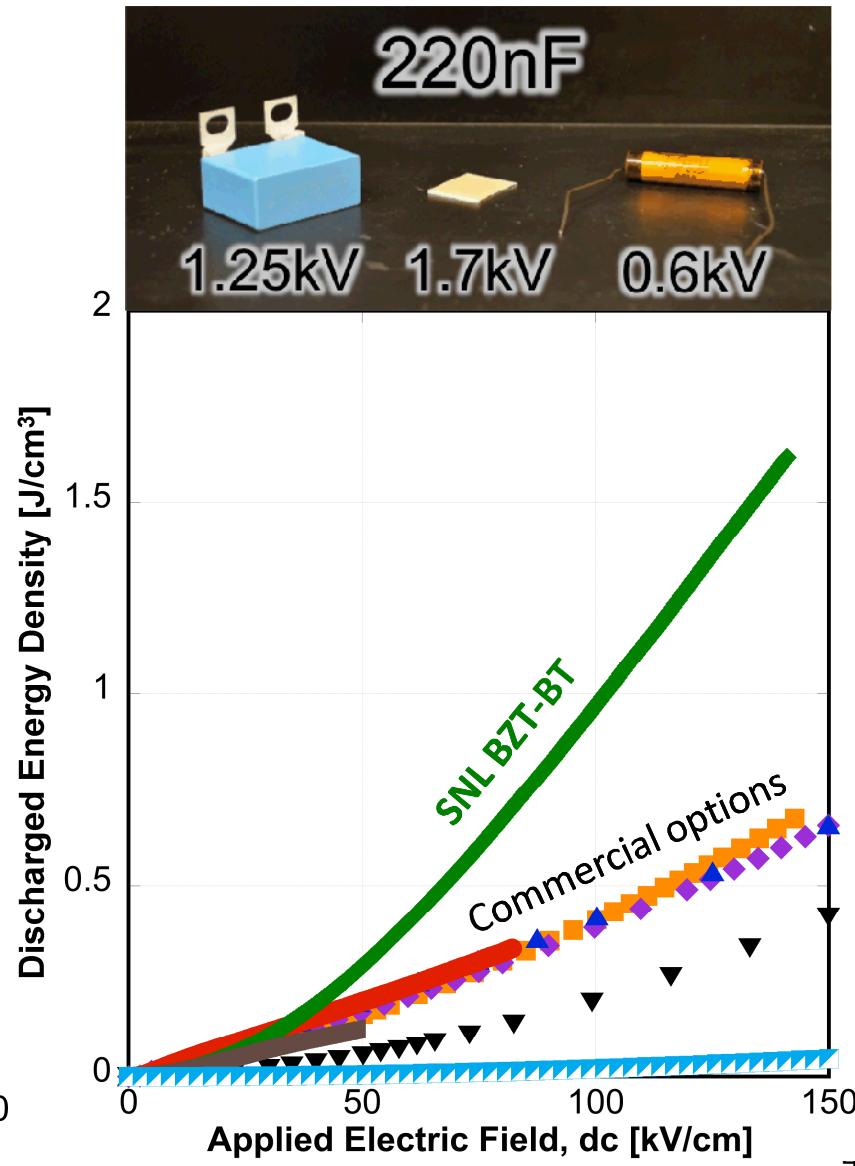
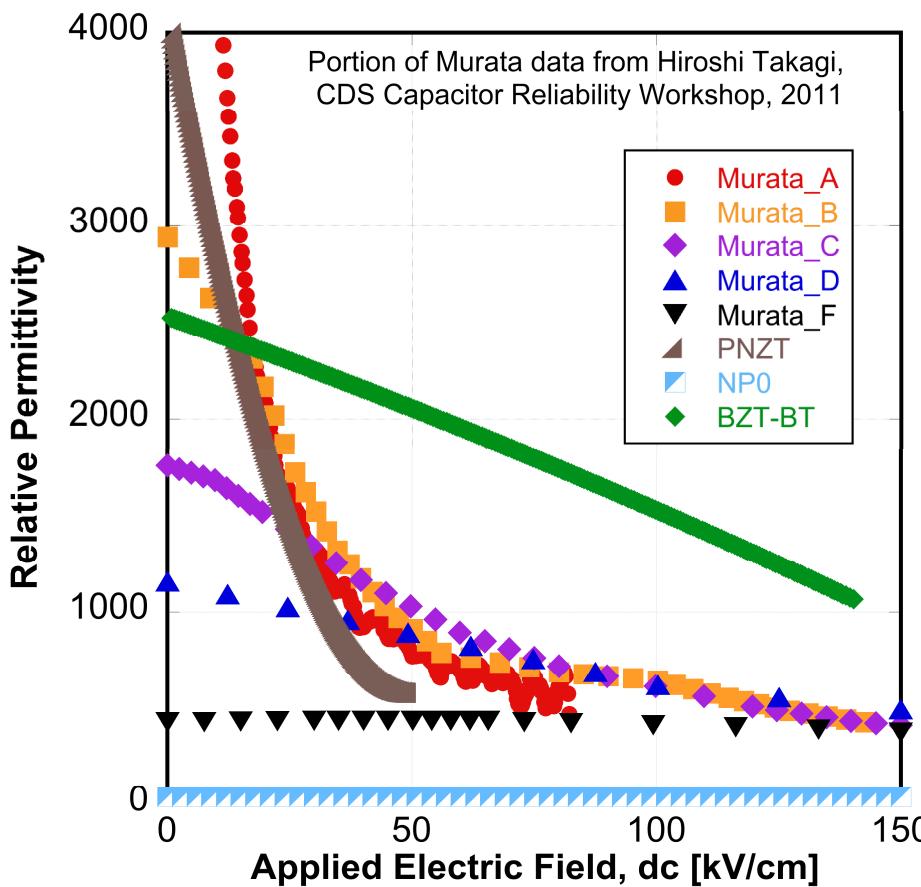
Pulse discharge applications require high energy density, low electromechanical response, low ESR



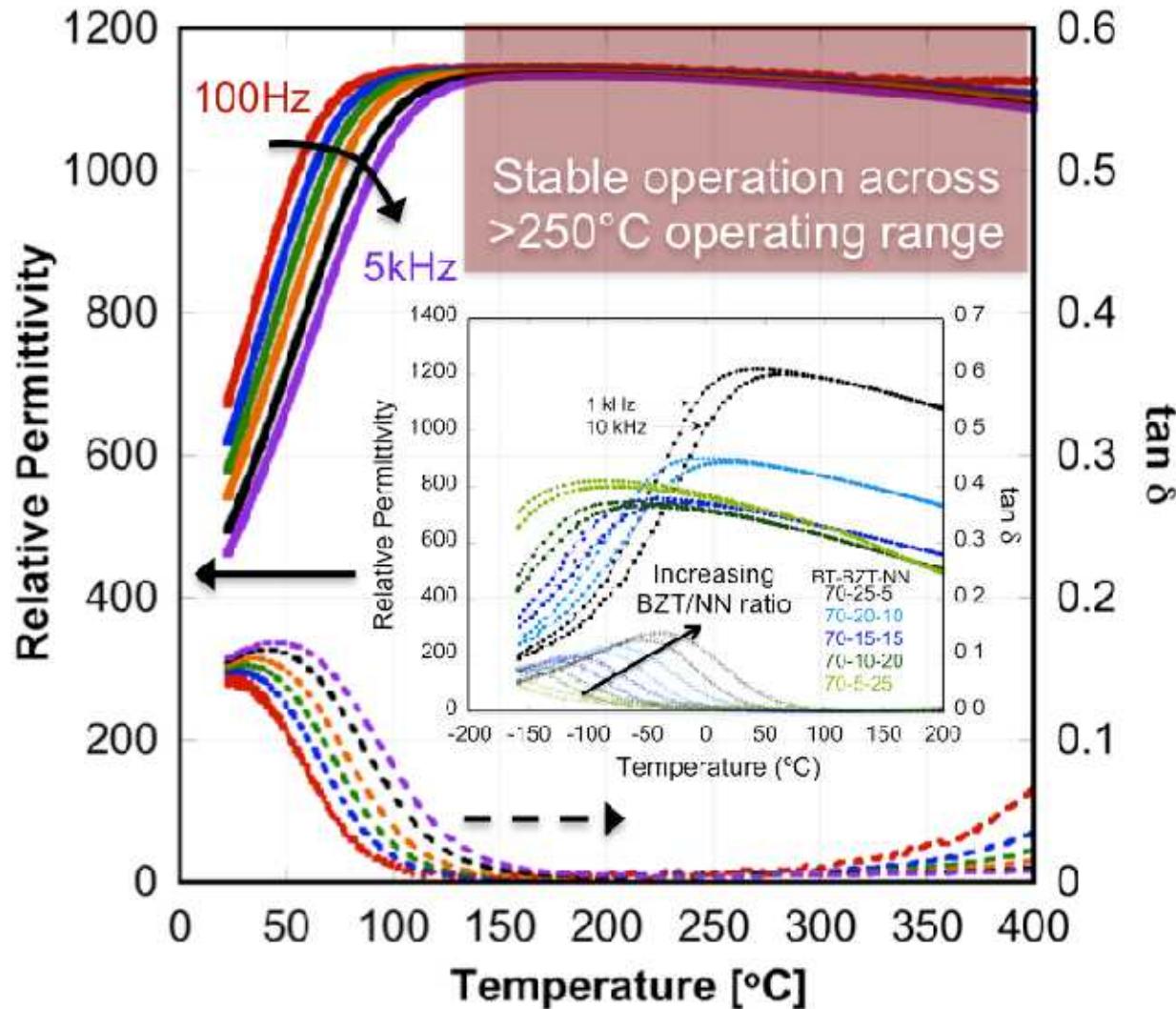
Simulation courtesy of Eric Green and Stan Atcitty

High Energy Density Dielectrics

$$J = \int_0^{V_{\max}} CV^2 \rightarrow \int_0^{E_{\max}} KE^2$$



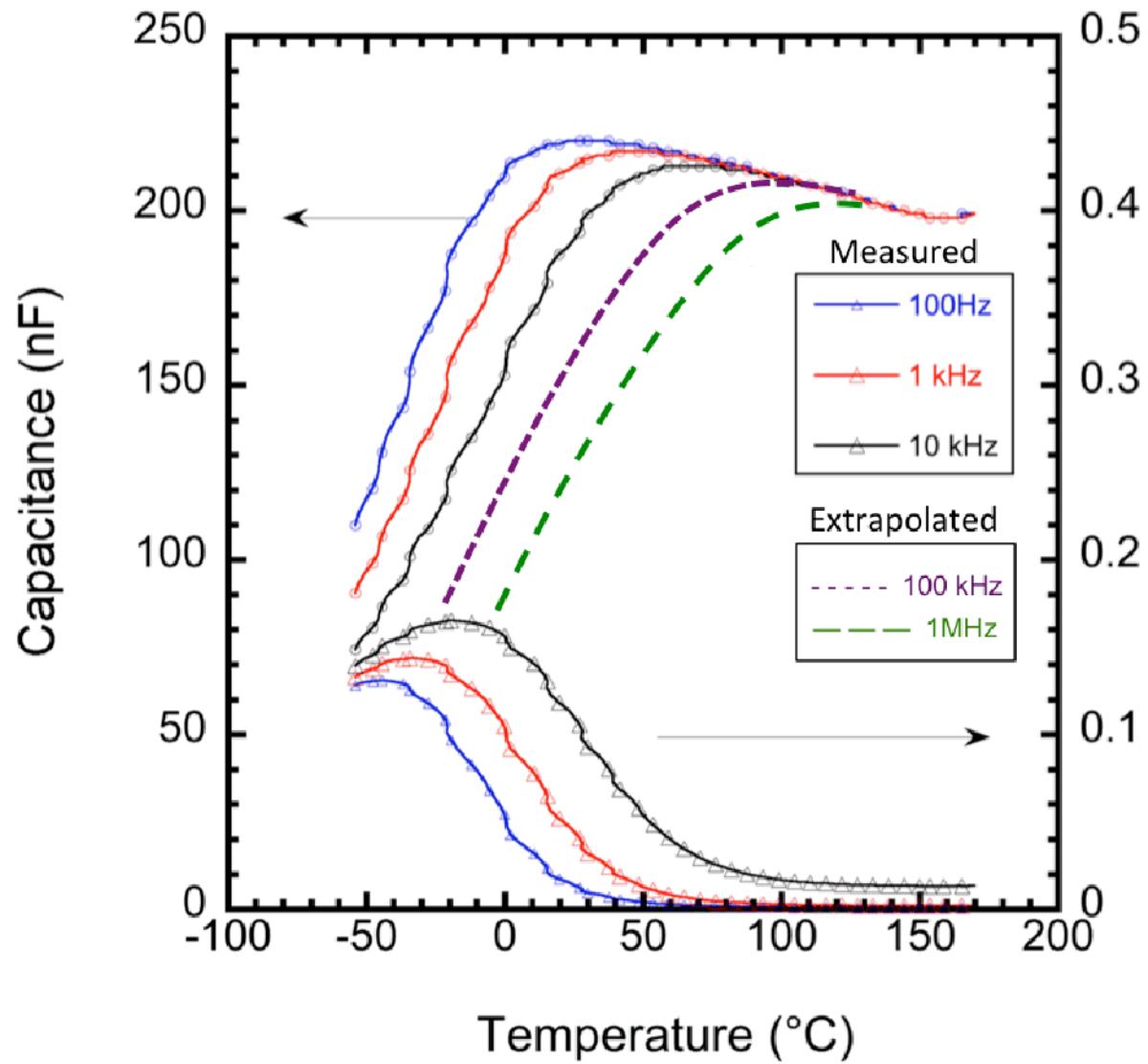
High Temperature Operation



- BiScO_3 stabilizes high temperature permittivity
- SrTiO_3 or NaNbO_3 additions shift relaxor transition to lower temperatures

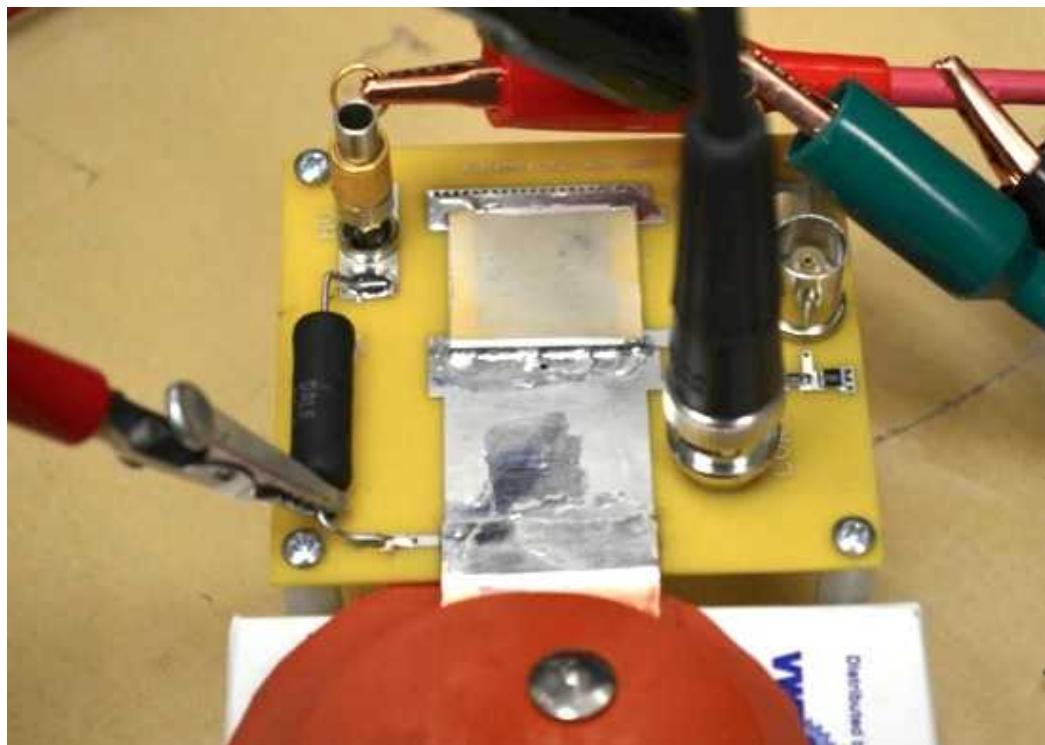
We can shift the temperature range for stable operation around by $\sim 250^\circ\text{C}$ via chemical modification *without sacrificing voltage stability*

Time Domain Performance



- Relaxor dielectrics exhibit characteristic frequency dispersion over relatively broad temperature ranges
- For pulse discharge or switched inverter applications that charge slowly and discharge quickly, which values are relevant?

Time Domain Performance



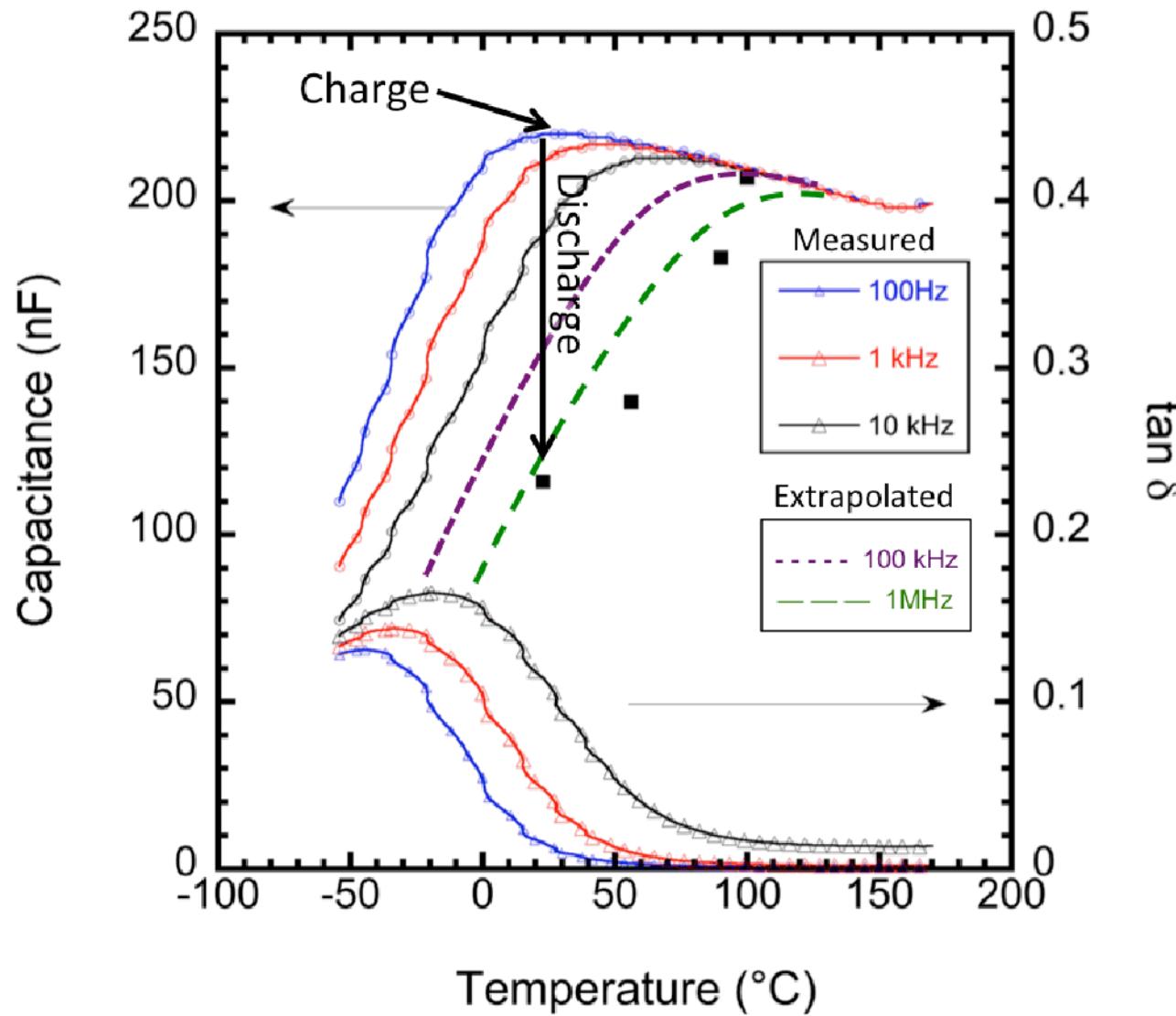
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Low inductance board with compensated and calibrated CVR for pulse discharge (time domain) measurements

Large piezoelectric response of traditional relaxor ferroelectric causes fracture during such testing

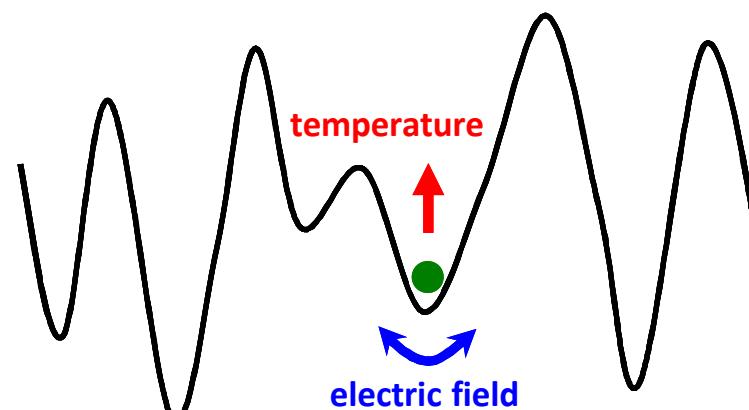
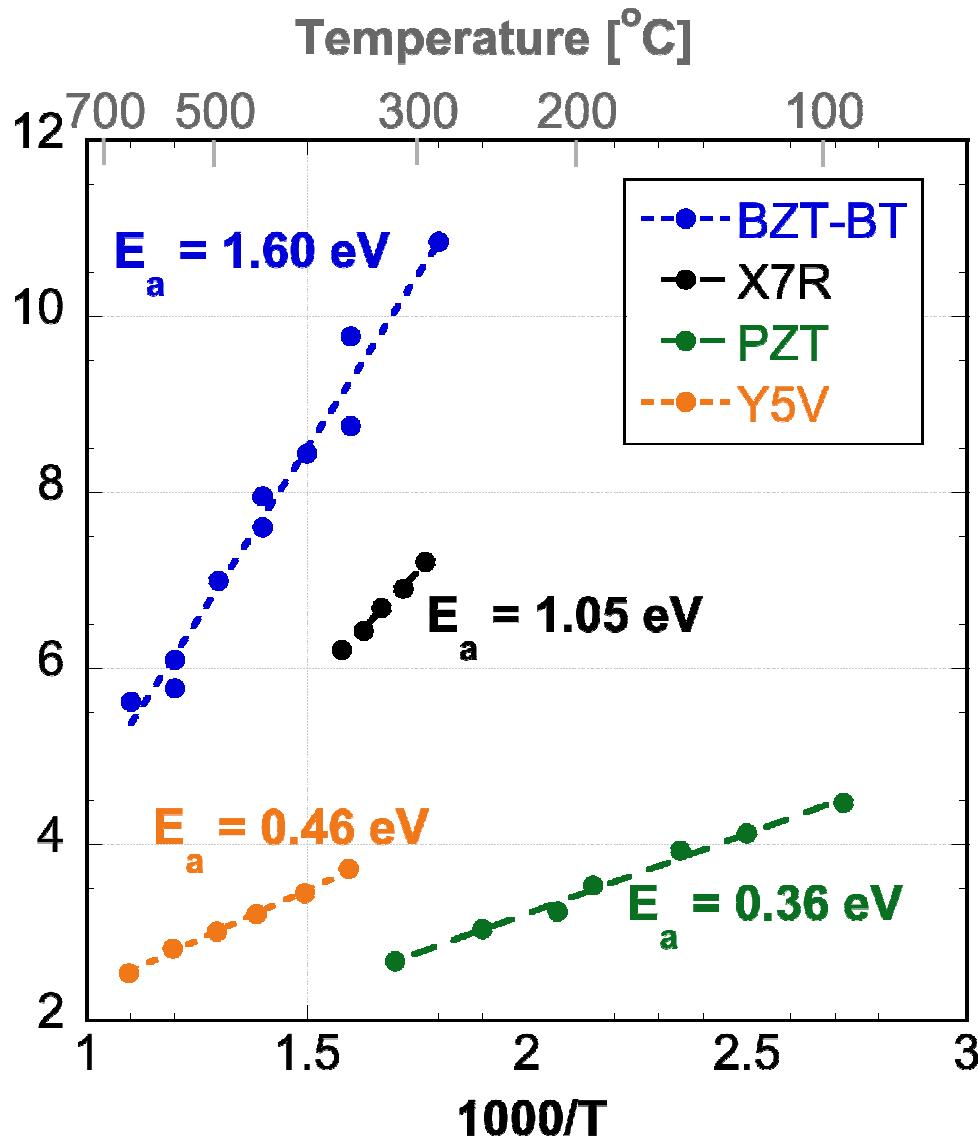
- Fitting critically damped discharge waveform allows determination of real part of permittivity

Time Domain Performance



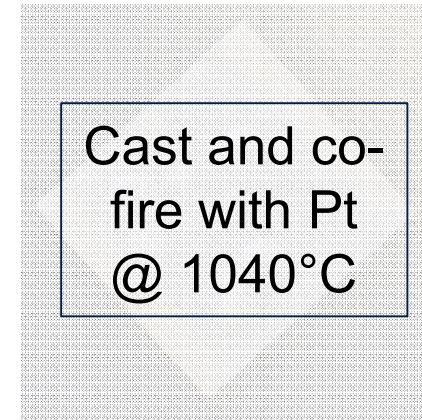
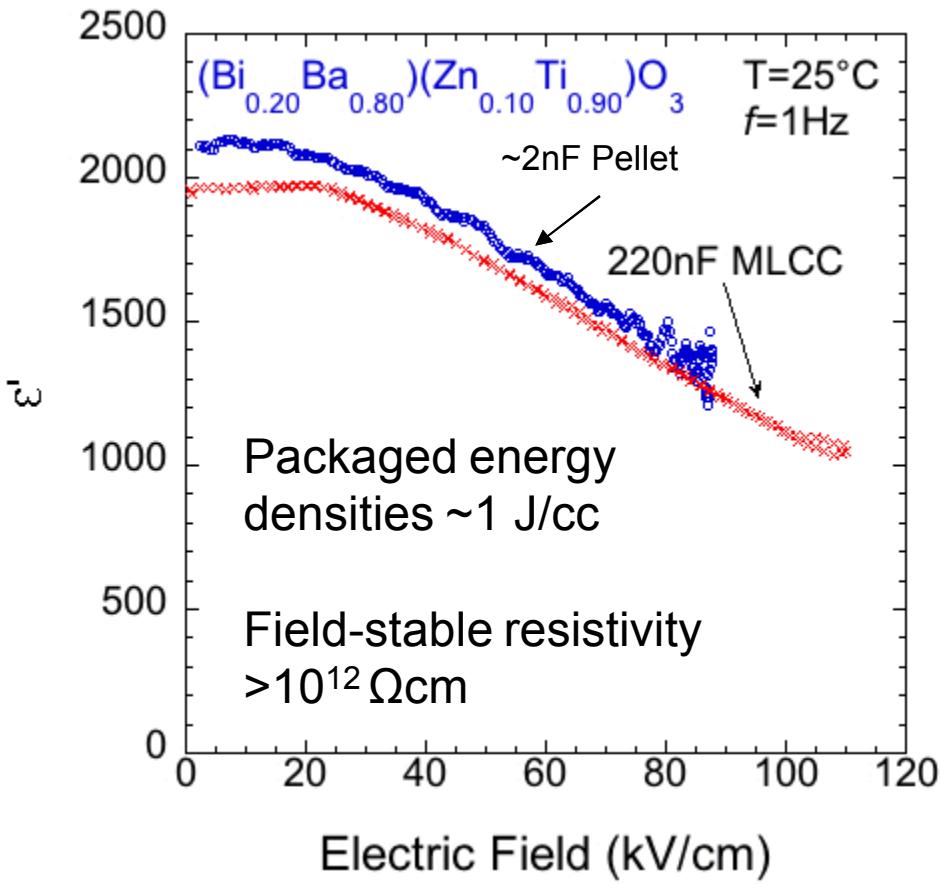
- Discharge data correspond to frequencies from 2.0-2.8 MHz
- Direct time domain results correlate well with extrapolated frequency-domain curves
- First known confirmation of relaxor response in time domain

High dc Resistivity → Reliable



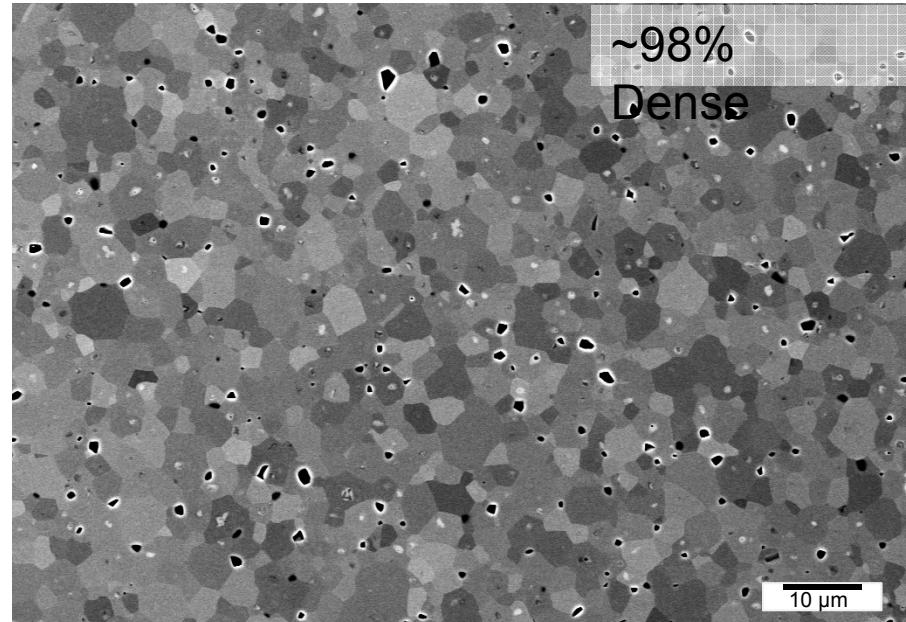
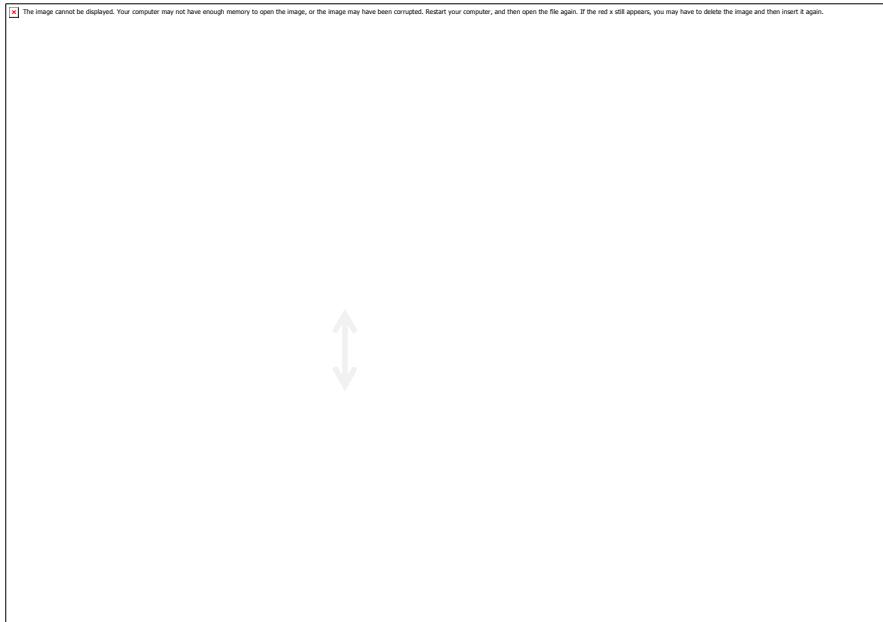
Higher resistivity and larger activation energy for conduction both translate into longer lifetimes and higher reliability, particularly at elevated temperatures.

Prototype MLCCs: 200nF @ 1700V

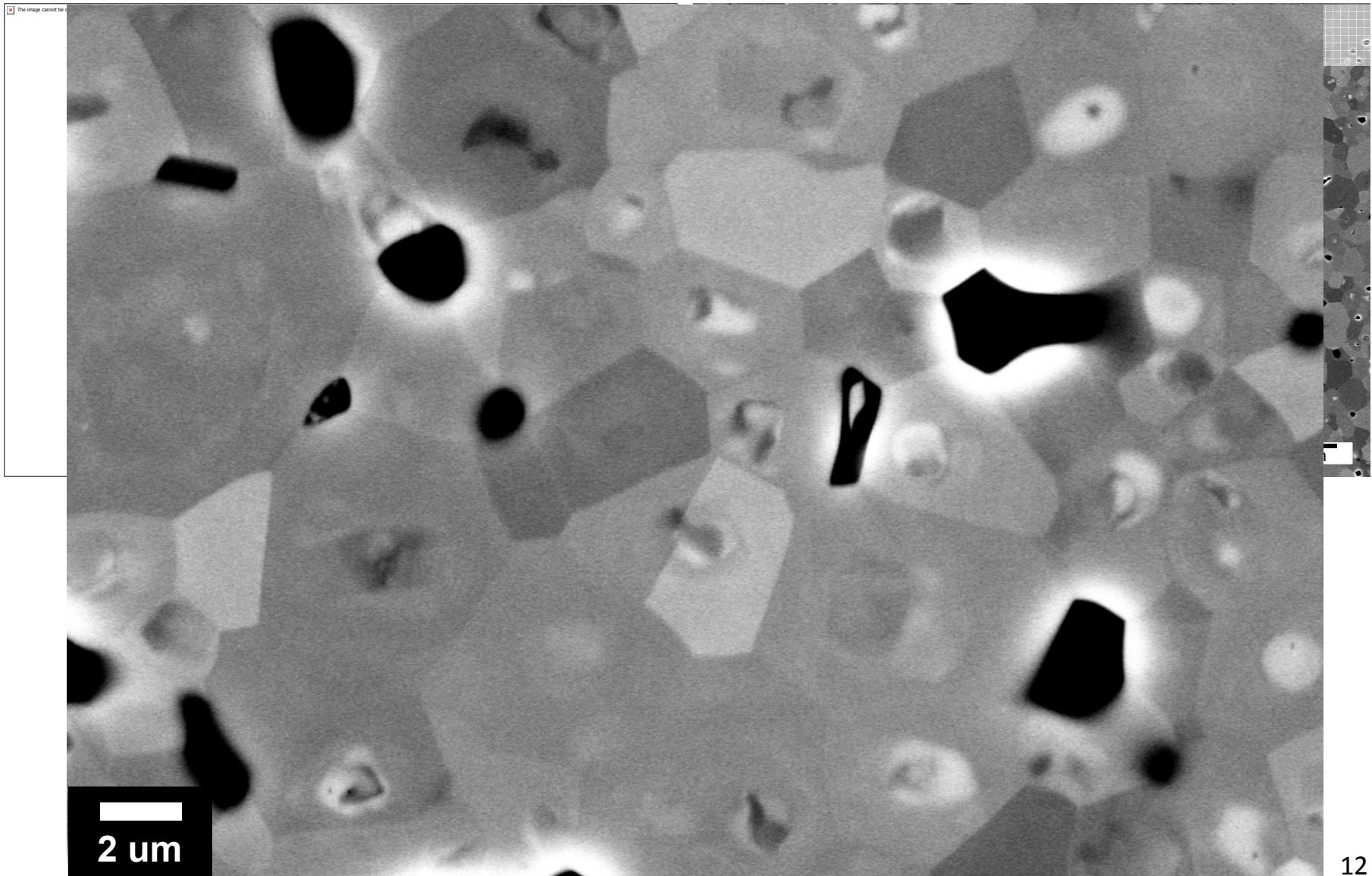


- Maintain RC time constants >200 sec even at $>60\text{kV/cm}$, $>150^\circ\text{C}$ (likely higher)
- Smaller samples tested under harsher conditions maintain RC >100 sec $>350^\circ\text{C}$
- Mechanism(s)??

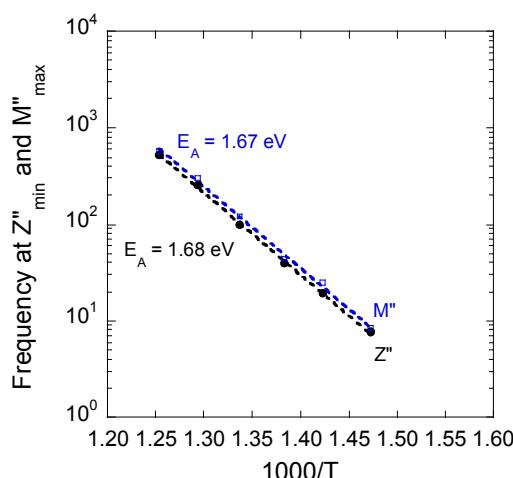
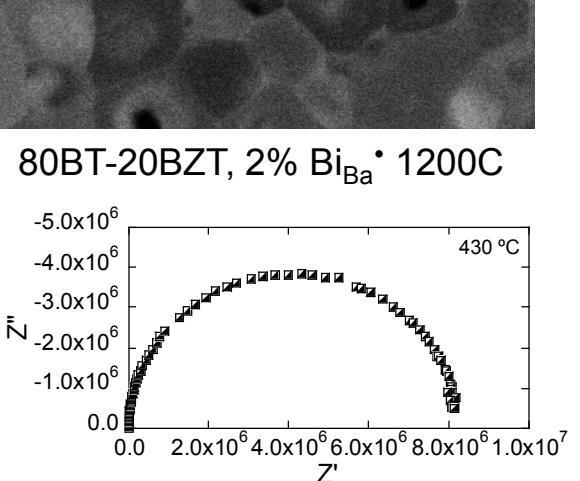
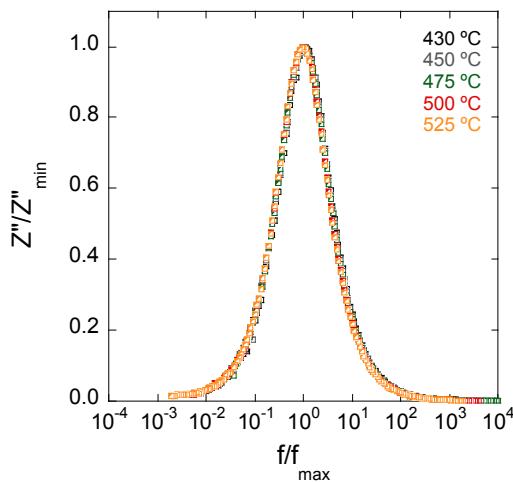
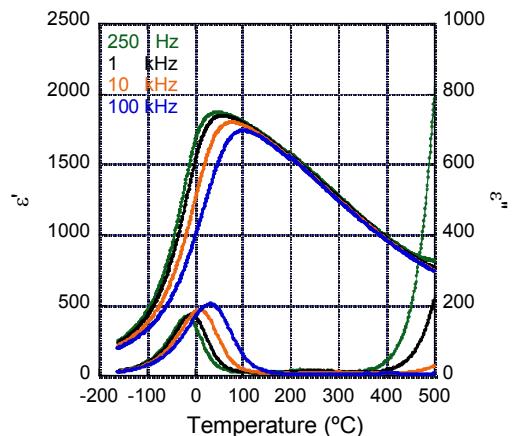
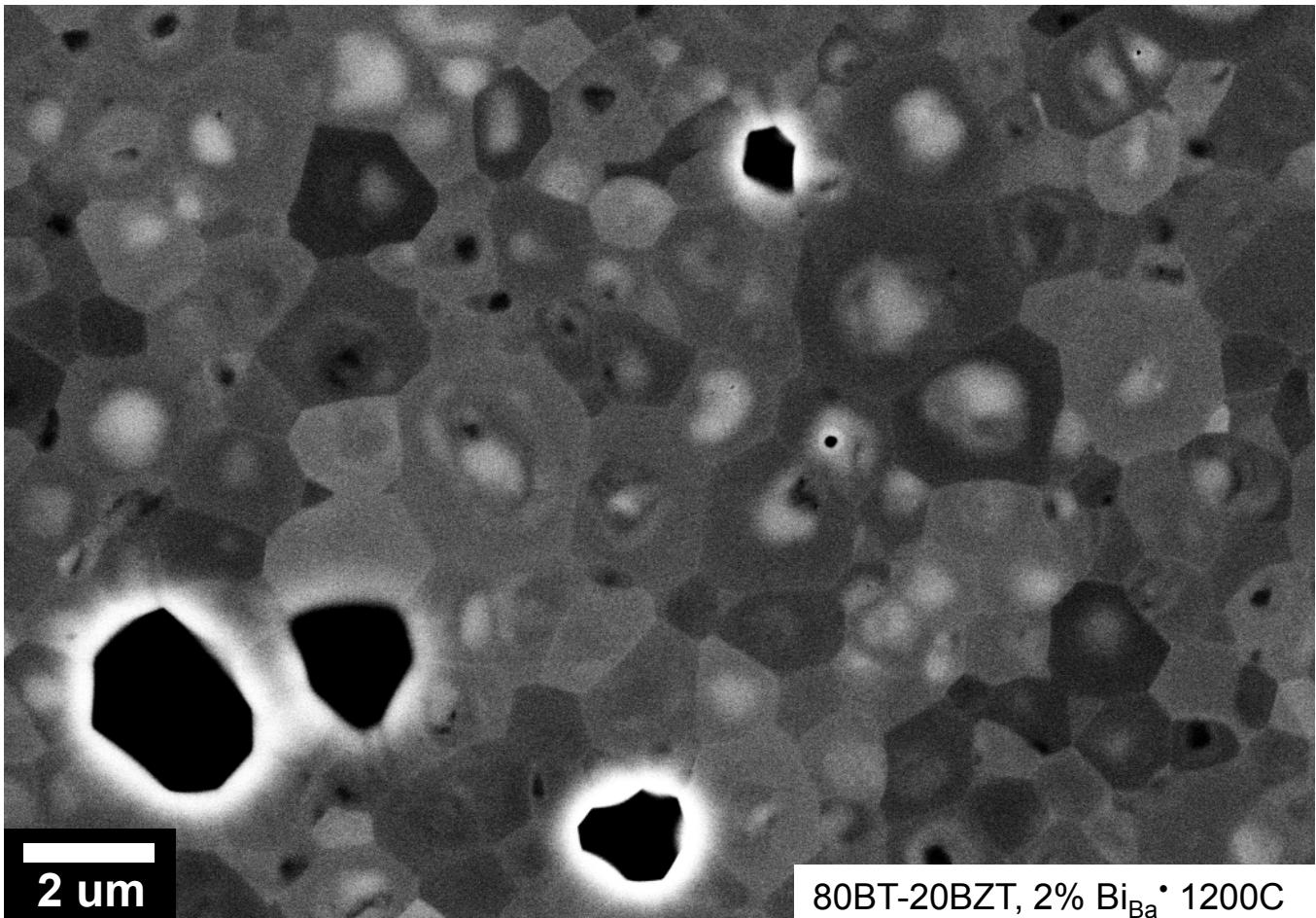
MLCC dielectric exhibits high-Z core



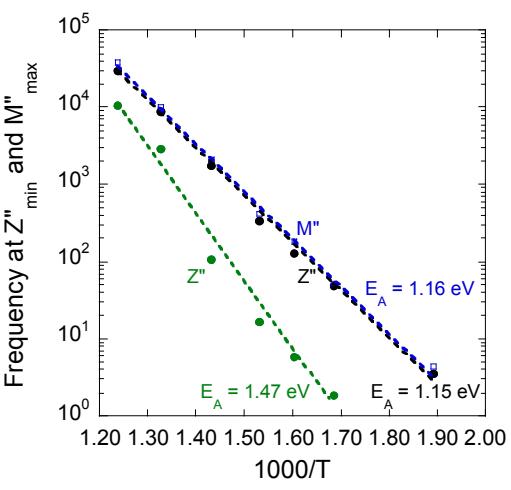
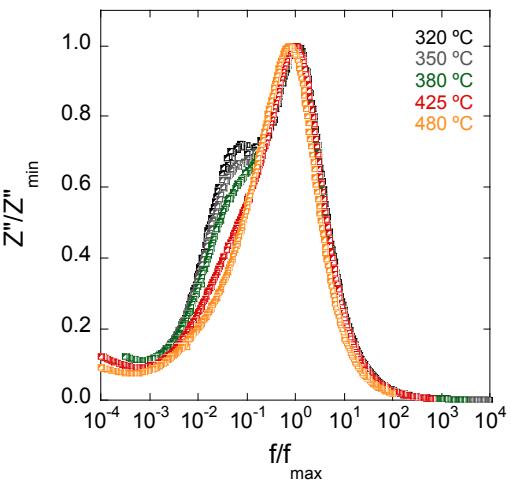
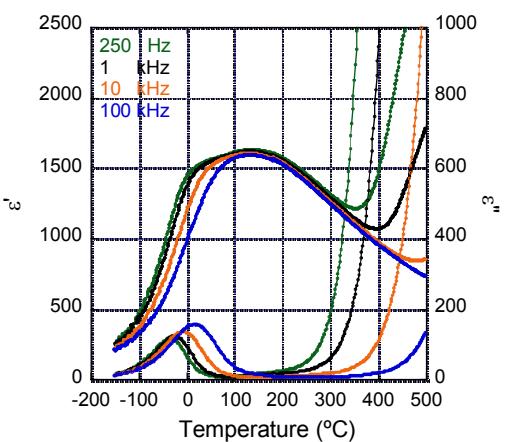
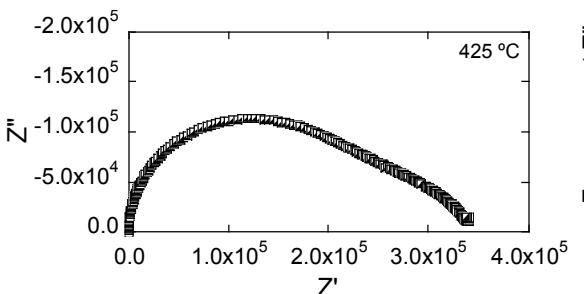
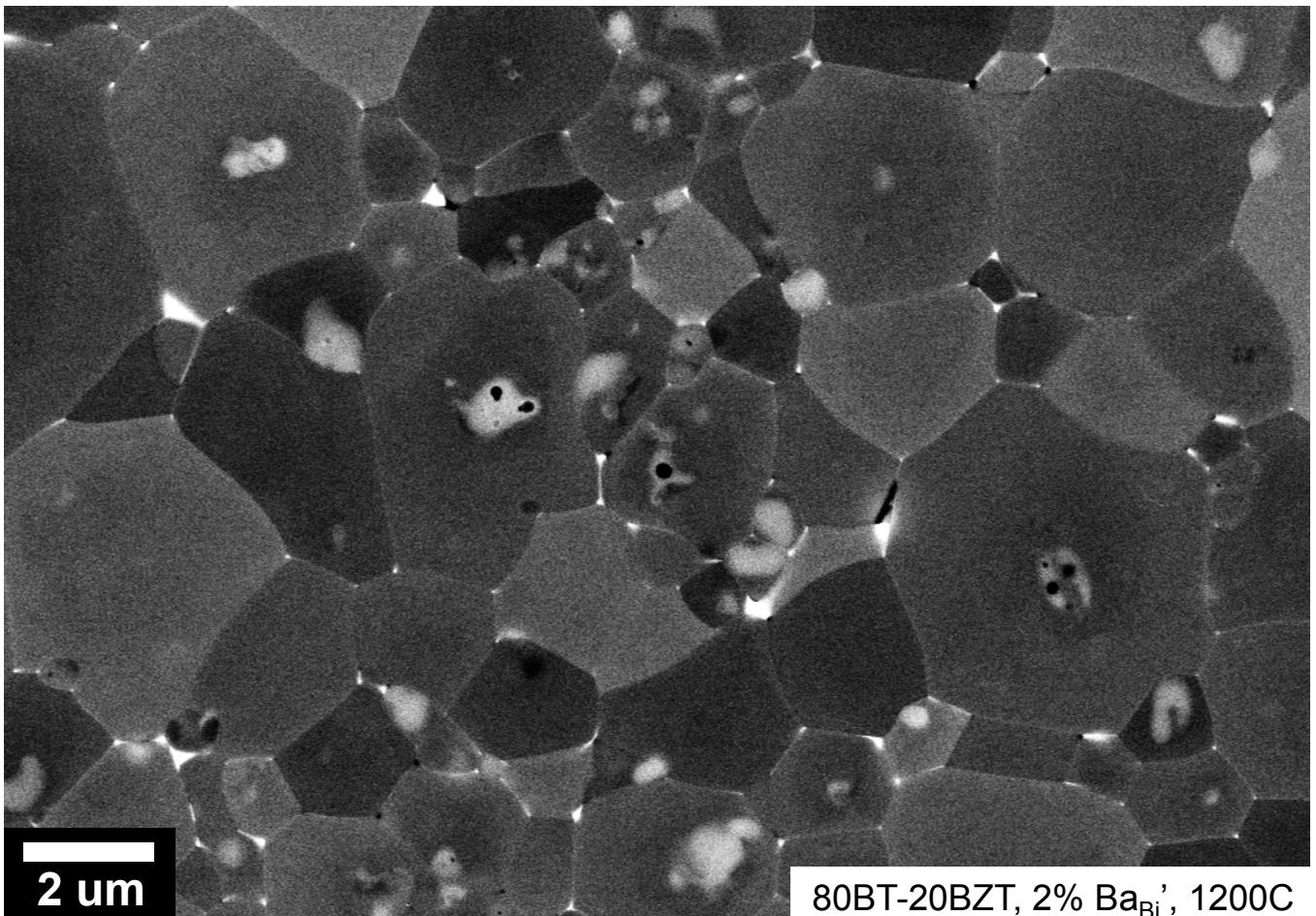
MLCC dielectric exhibits high-Z core



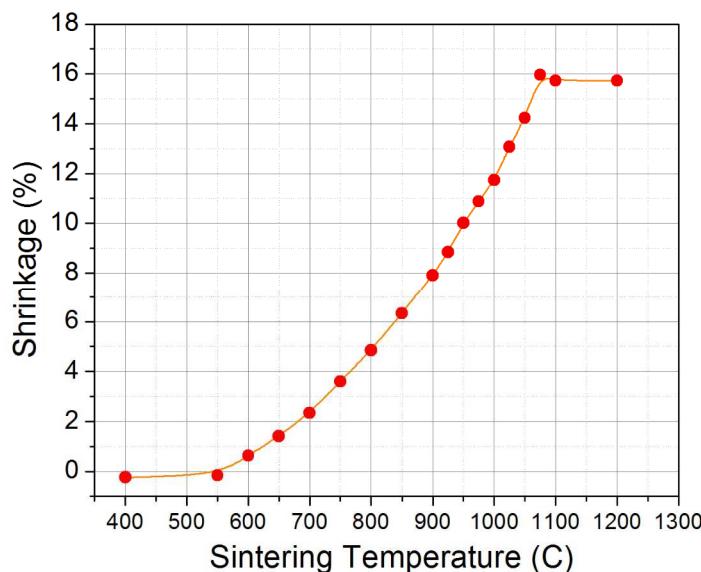
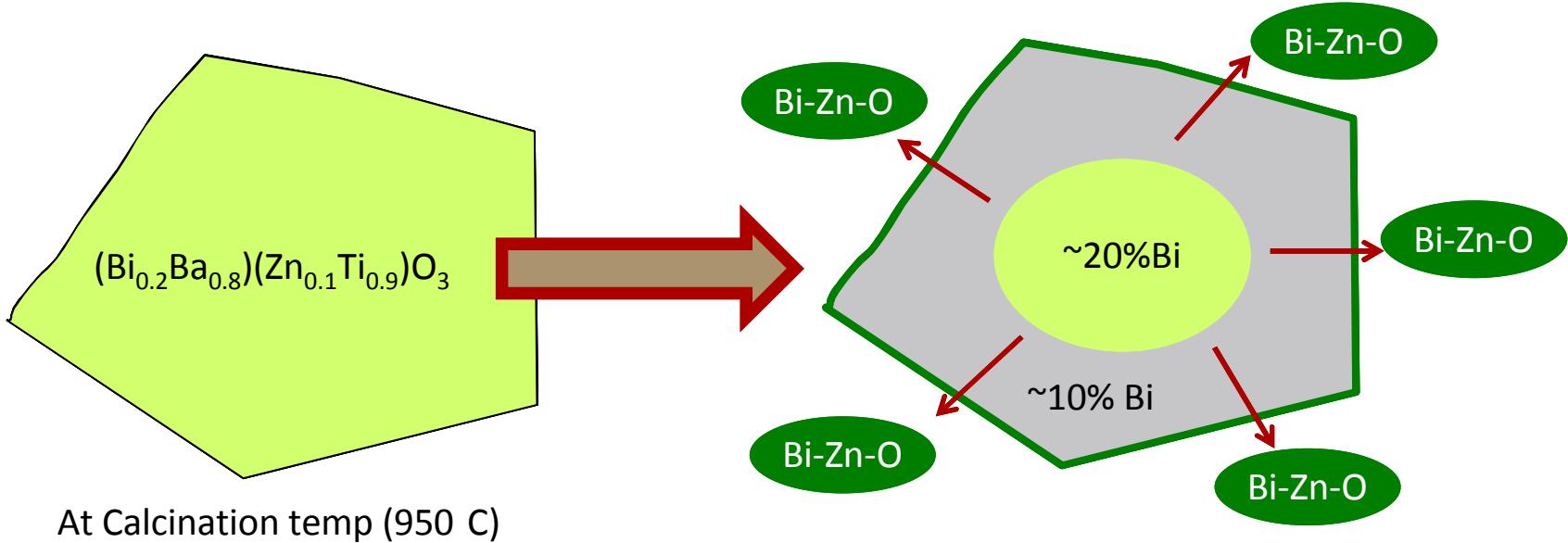
Donor-doped



Acceptor-doped



Calcination

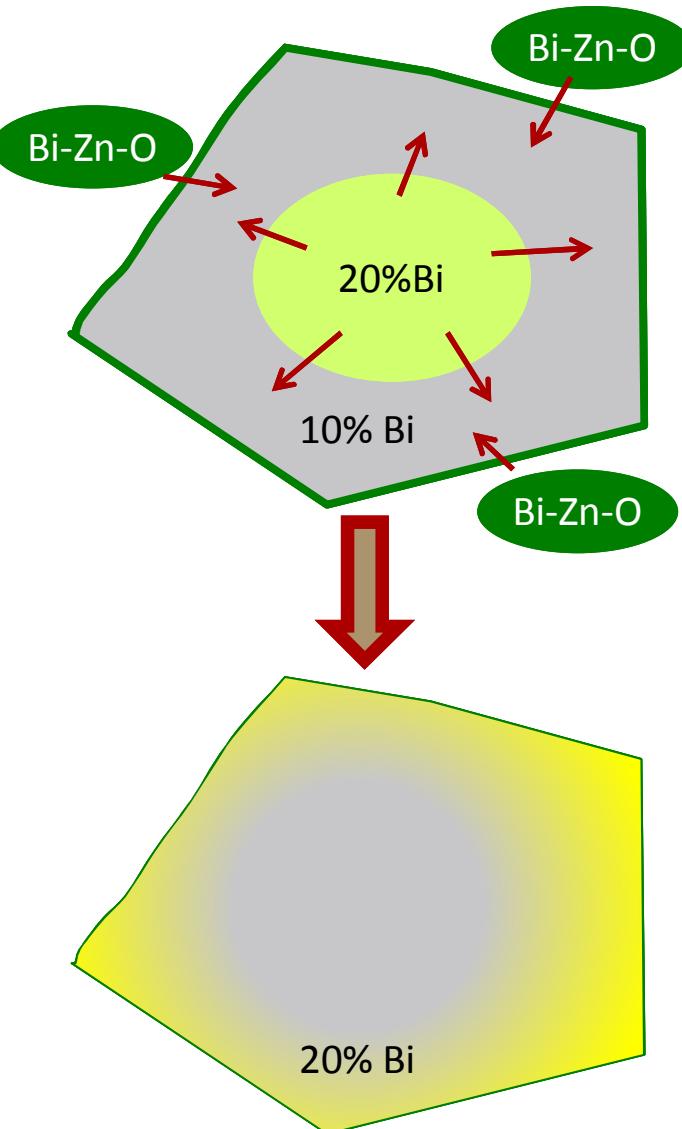


Working assumption is that the equilibrium solubility of Bi_2O_3 in BaTiO_3 is breached upon cooling.

On cooling from 950 C, Bi exsolves from perovskite solid solution, leading to Bi-rich core and Bi-Zn-O glassy phase at particle surfaces.

Initial sintering occurs as Bi-Zn-O phase melts. Next step depends on **dominant defect species**.

Sintering: Donor doped



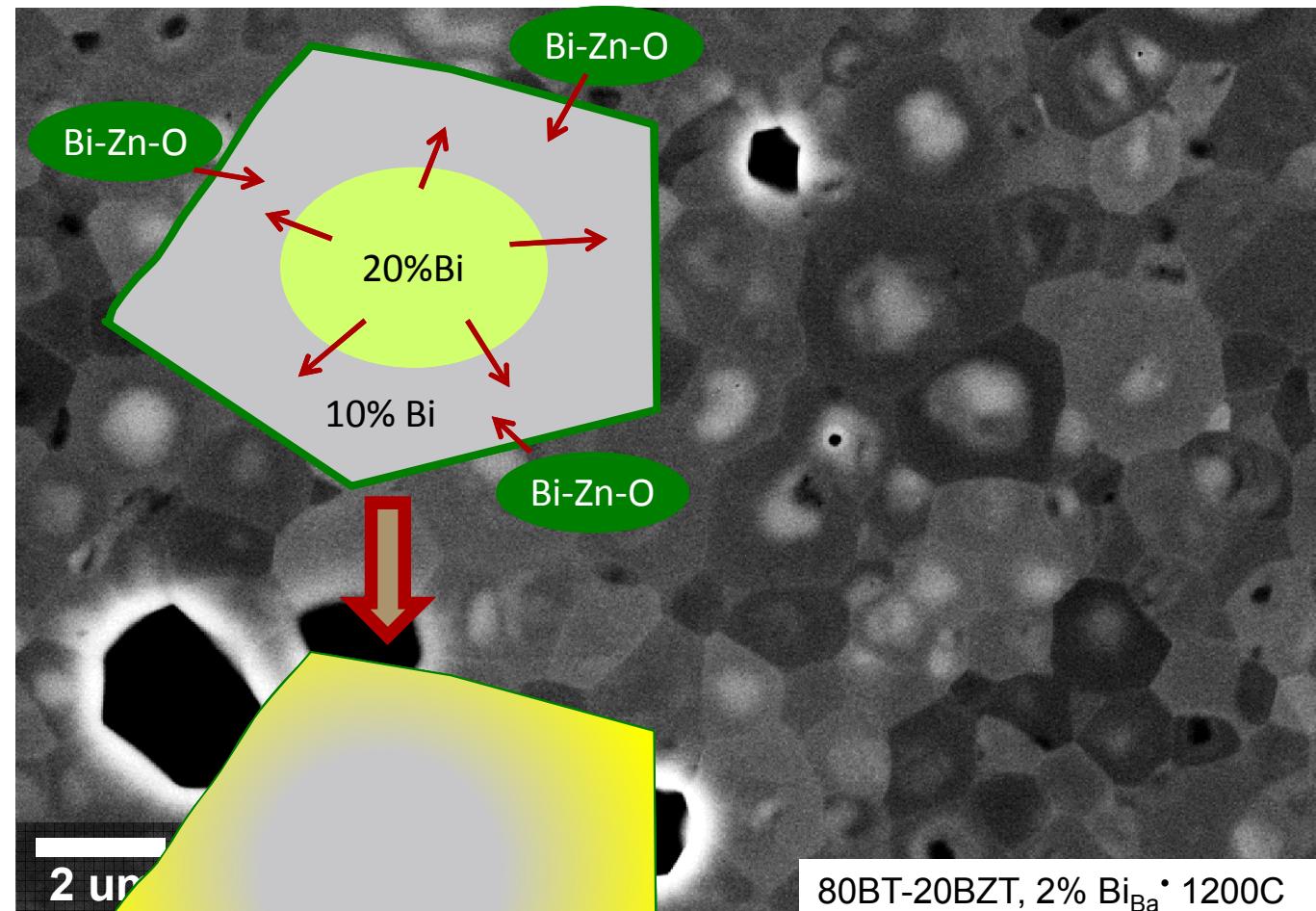
Donor Doped: Cation Vacancies Dominate

- Bi diffuses in from g.b., out from core
- Diffusion **assisted** by cation vacancies

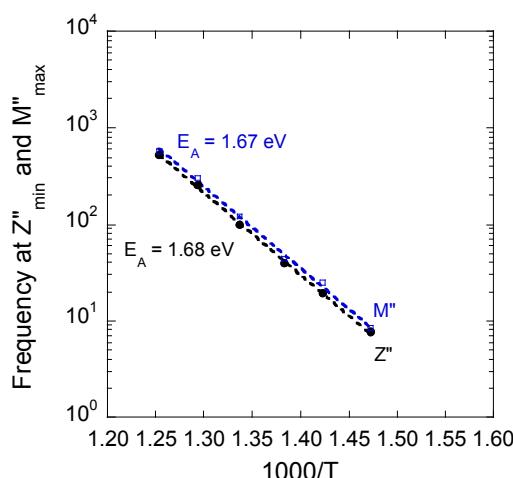
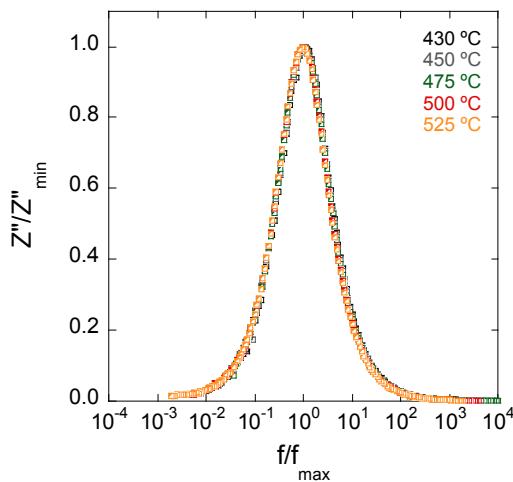
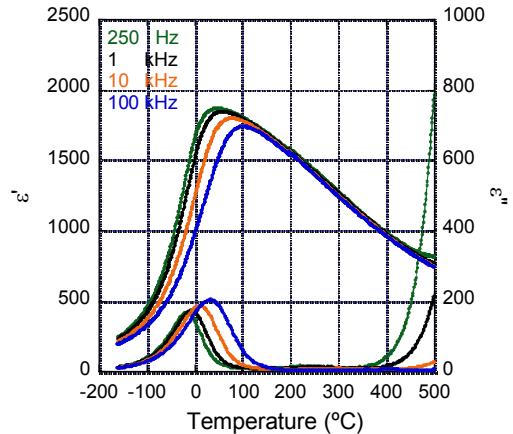
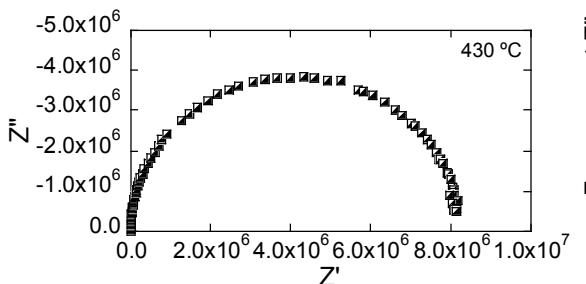
Resultant Microstructure

- Relatively homogenous
- Bi-gradient with diffuse boundary due to ample Bi diffusion
- Electrically homogeneous microstructure (single relaxation)

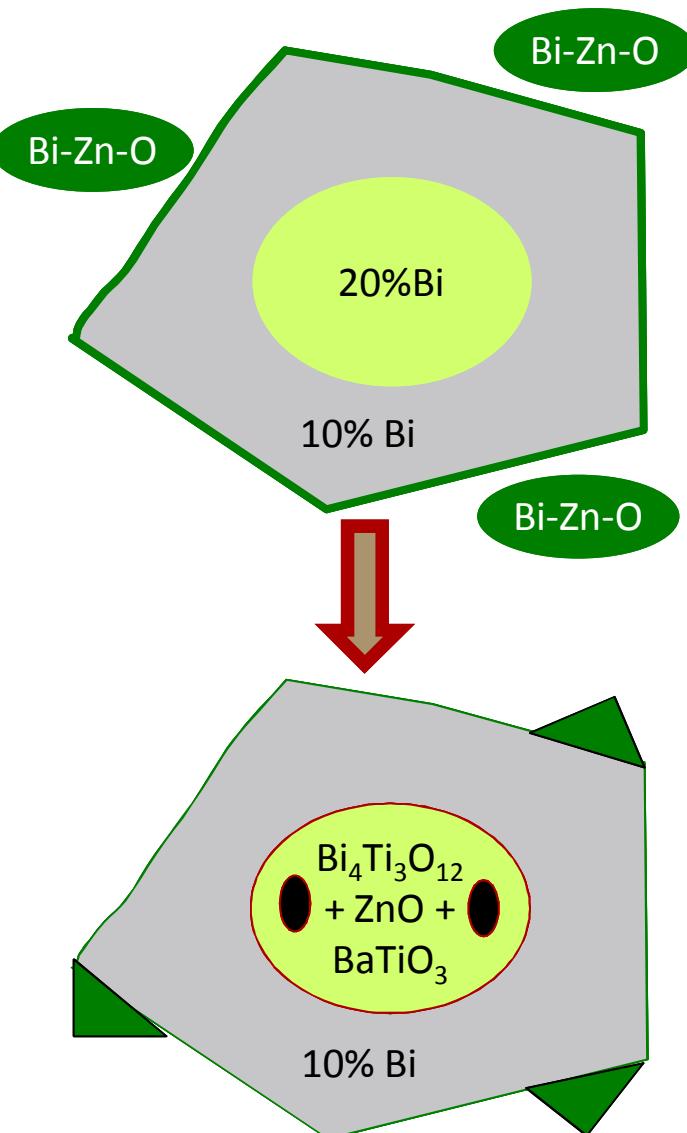
Sintering: Donor doped



80BT-20BZT, 2% Bi_{Ba} • 1200C



Sintering: Acceptor doped



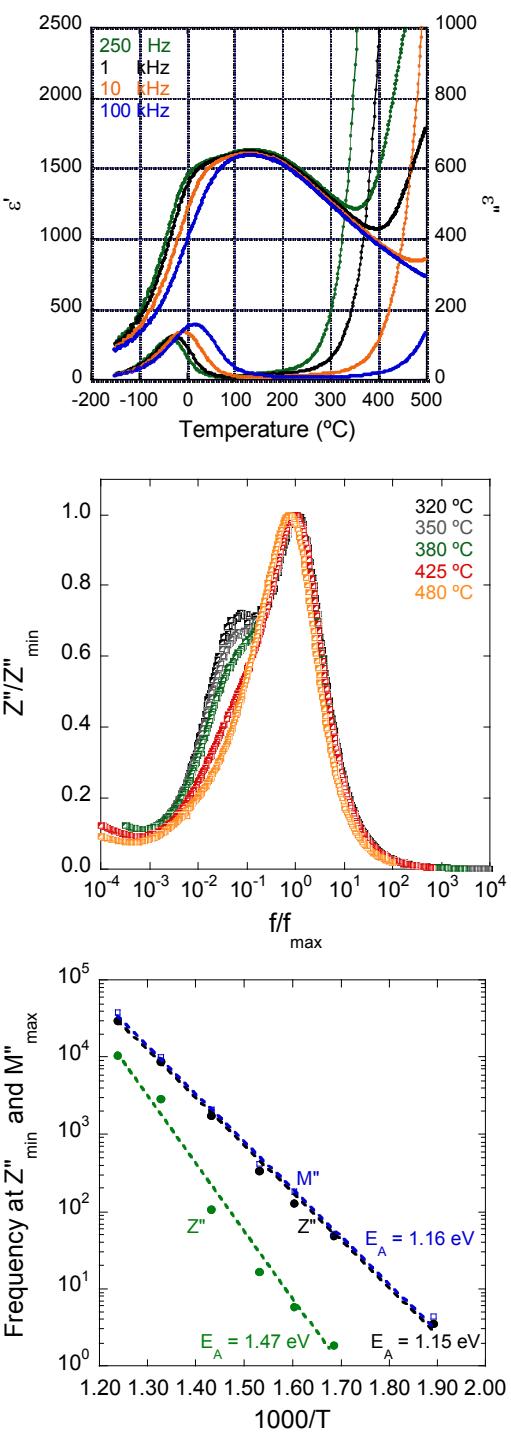
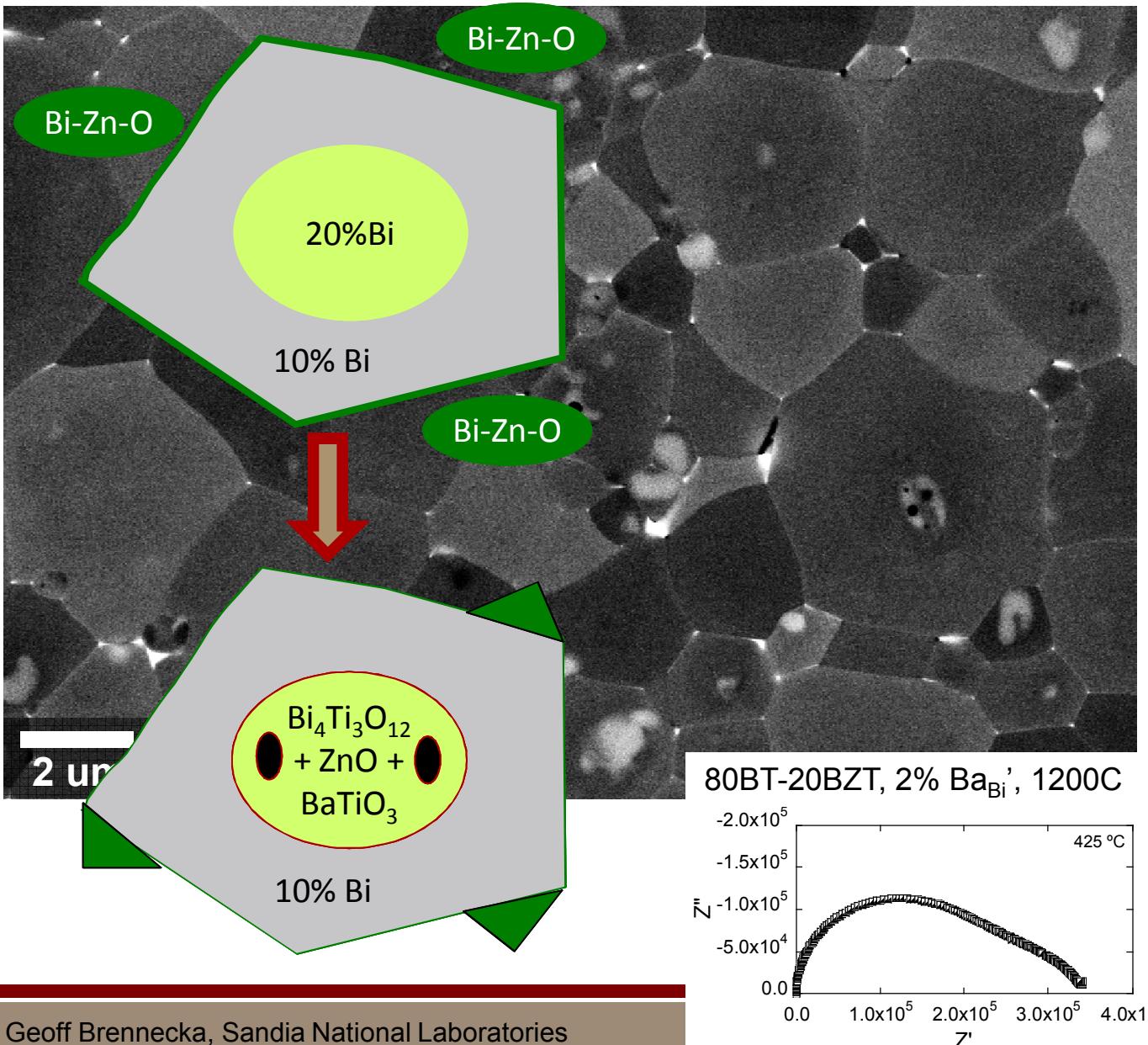
Acceptor Doped: Oxygen Vacancies Dominate

- Bi diffuses in from g.b., out from core
- Diffusion **inhibited** by lack of cation vacancies
- Bi-rich phase at triple points
- Bi-rich cores decompose into equilibrium phases: $\text{Bi}_4\text{Ti}_3\text{O}_{12} + \text{ZnO} + \text{Ba-Ti-O}$

Resultant Microstructure

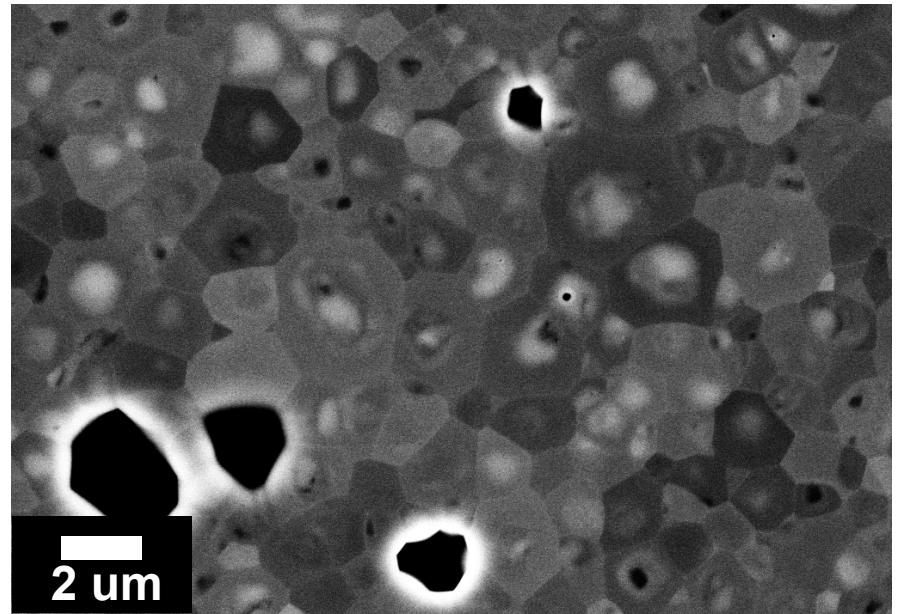
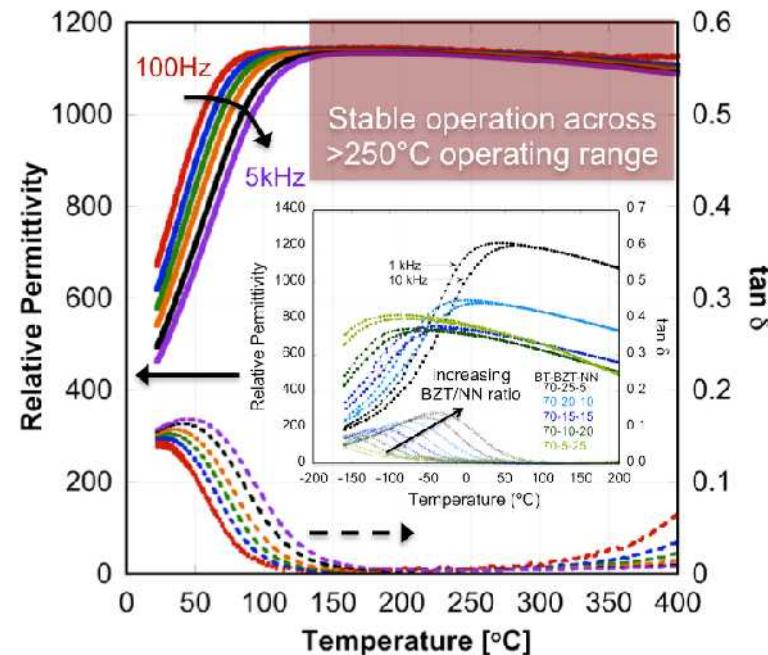
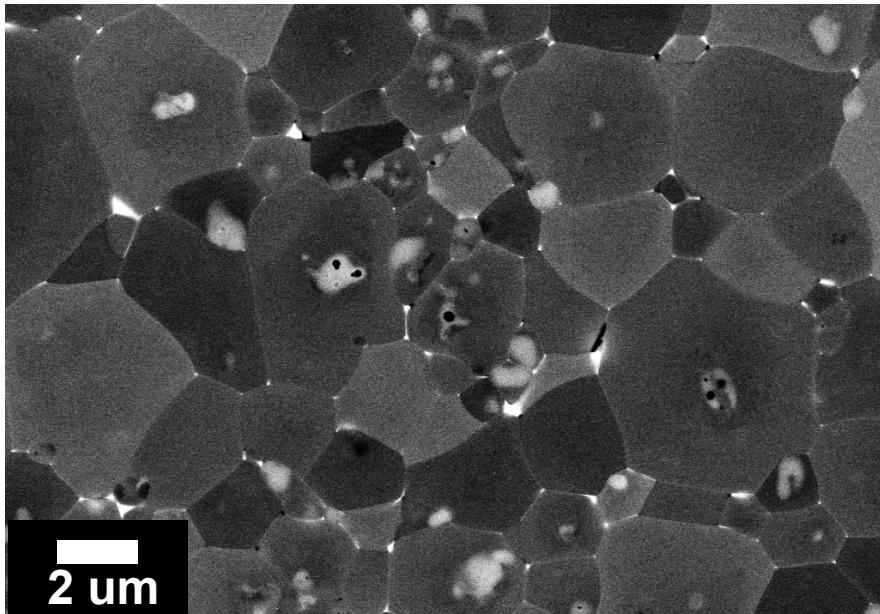
- High-Z phase at triple points
- Well defined core boundary
- Low-Z precipitates in core region
- Electrically heterogeneous microstructure (two relaxations)

Sintering: Acceptor doped

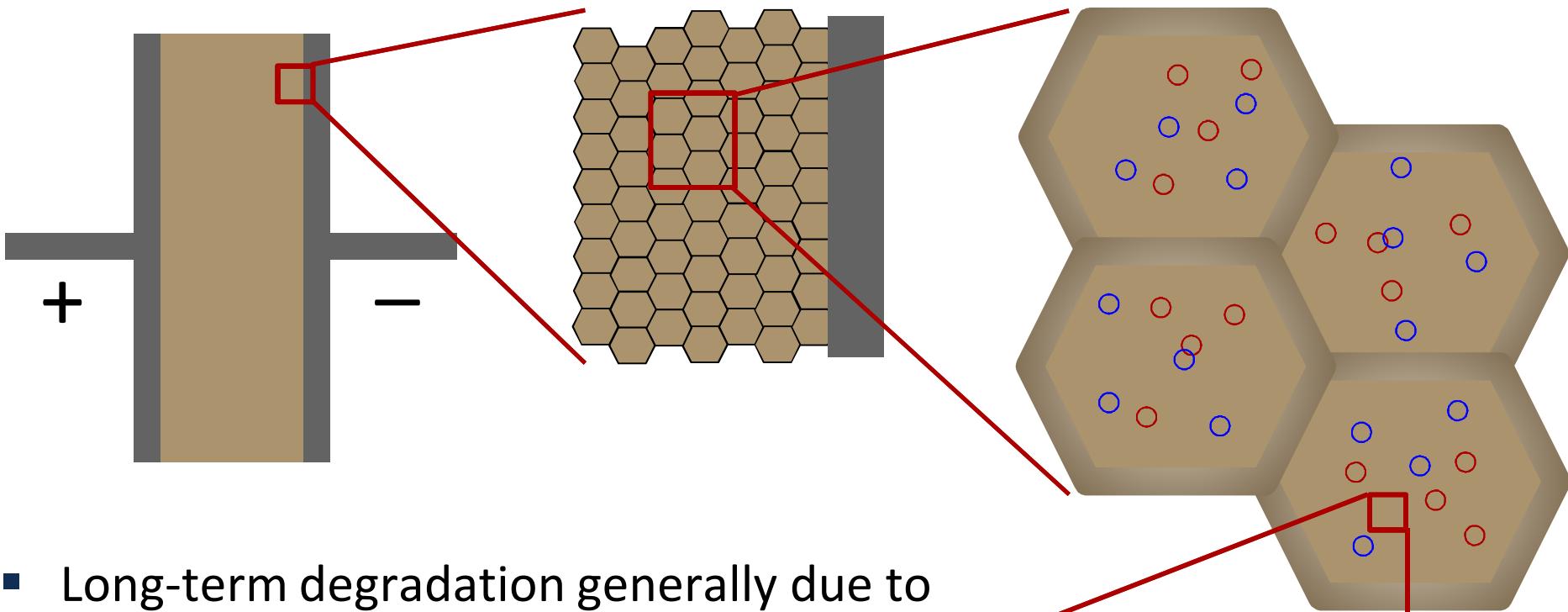


Summary

- Fabricated high energy density dielectrics with temperature- and voltage-stable performance
- Defect chemistry has strong influence on microstructure as well as electrical response



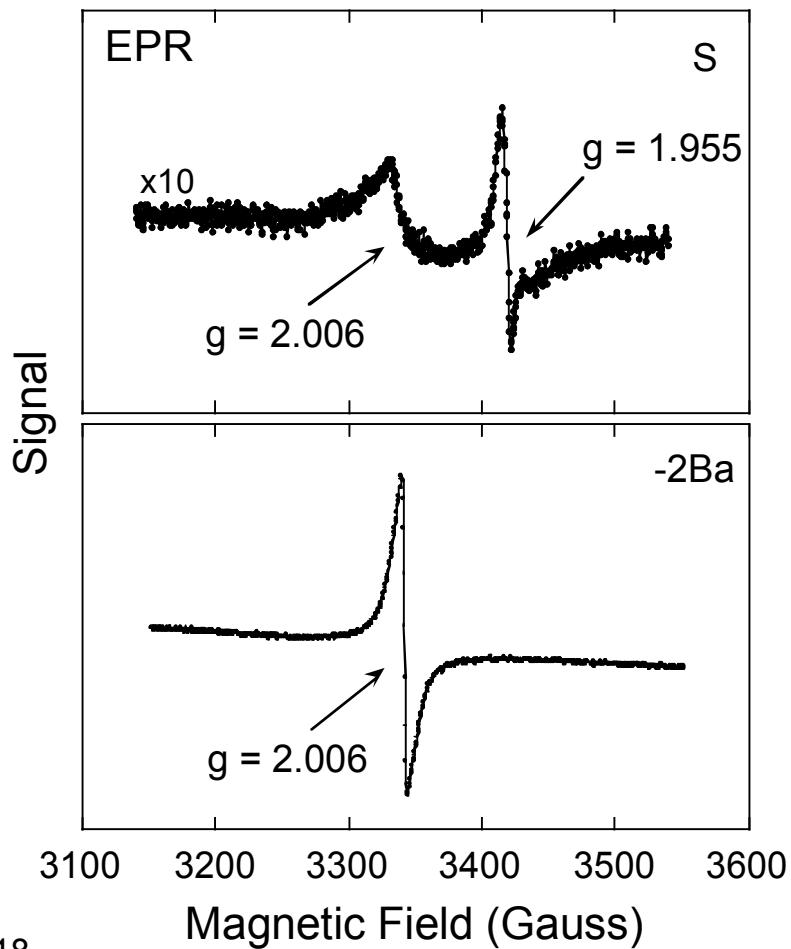
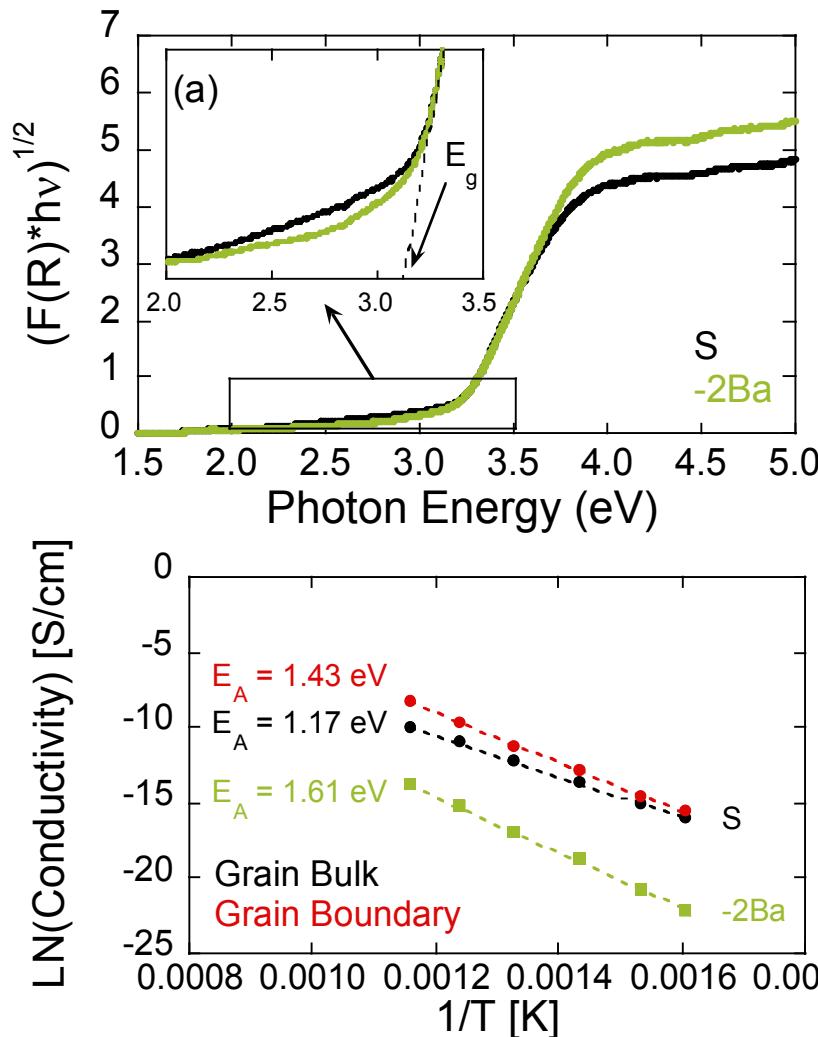
Degradation in Ceramic Dielectrics



- Long-term degradation generally due to migration of charged ionic defects
- HALT and MTTF prediction require quantitative description of mechanisms

For excellent discussion, see Randall *et al.*, *J. Appl. Phys.* **113**, 014101 (2013) doi: 10.1063/1.4772599

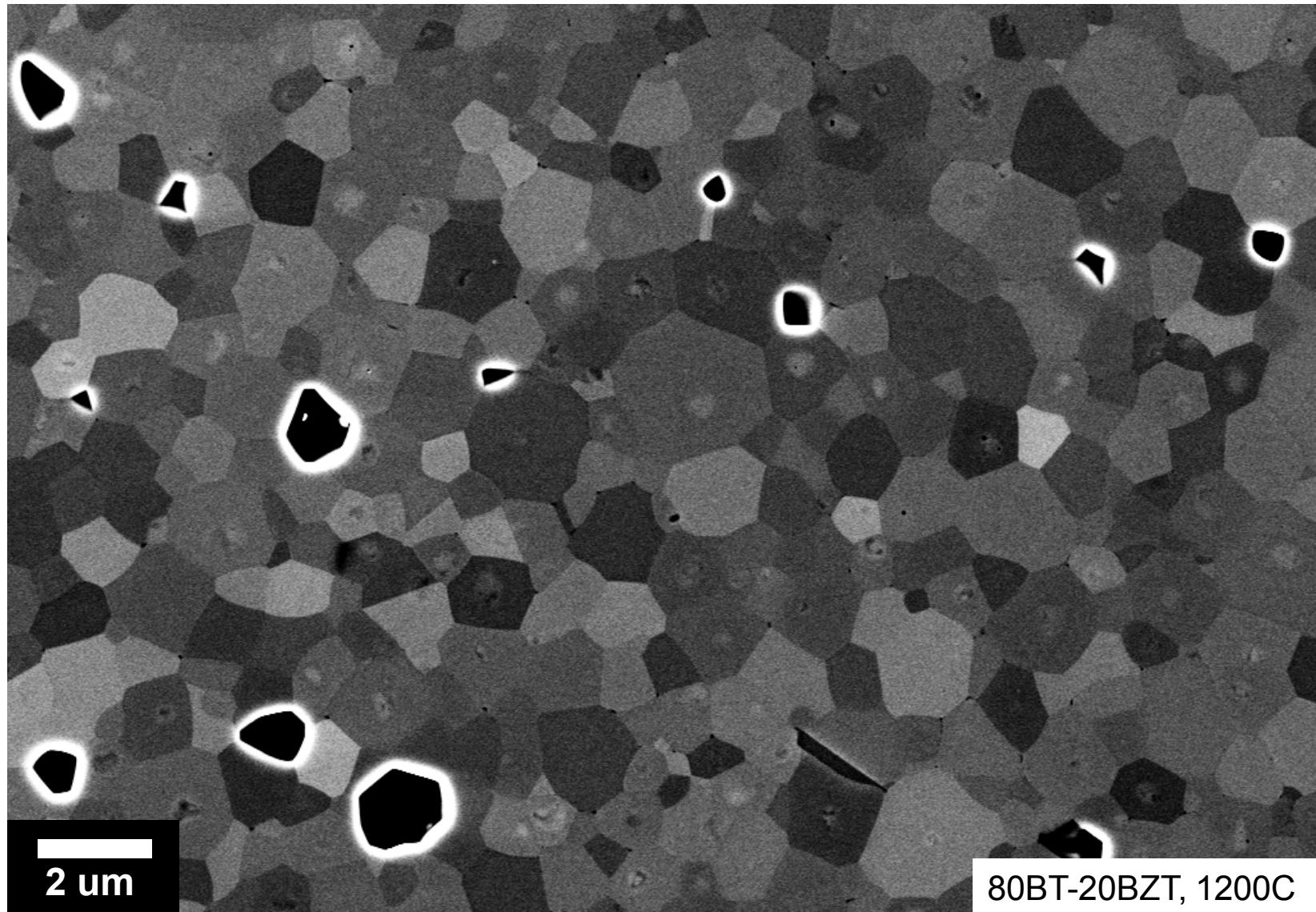
Defects: Optical & Electrical Info



Raengthon, et al., Appl. Phys. Lett. (2012)

- Defect studies suggest $V_{\text{Ba}}^{''} - V_{\text{O}}^{00}$ pairs are strong carrier traps

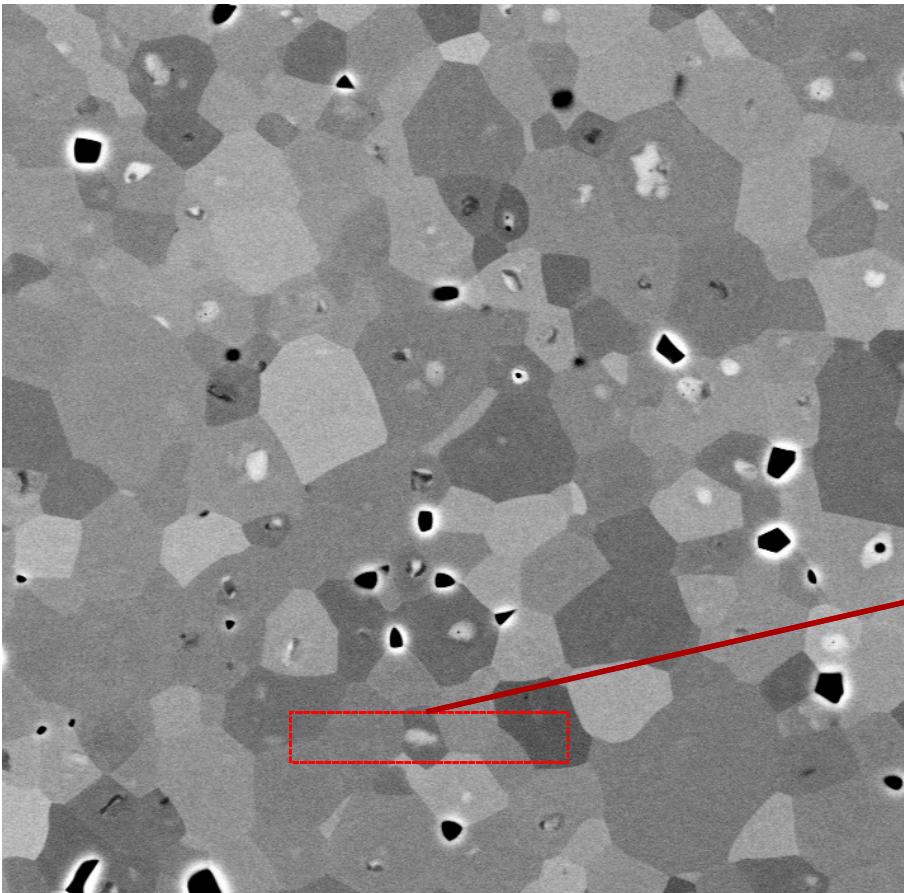
Well-Polished Section, Channeling



Compositional Variation

SEM

HAADF TEM

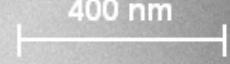


EHT = 15.00 kV WD = 5.5 mm Signal A = BSD File Name = MLCC_1A

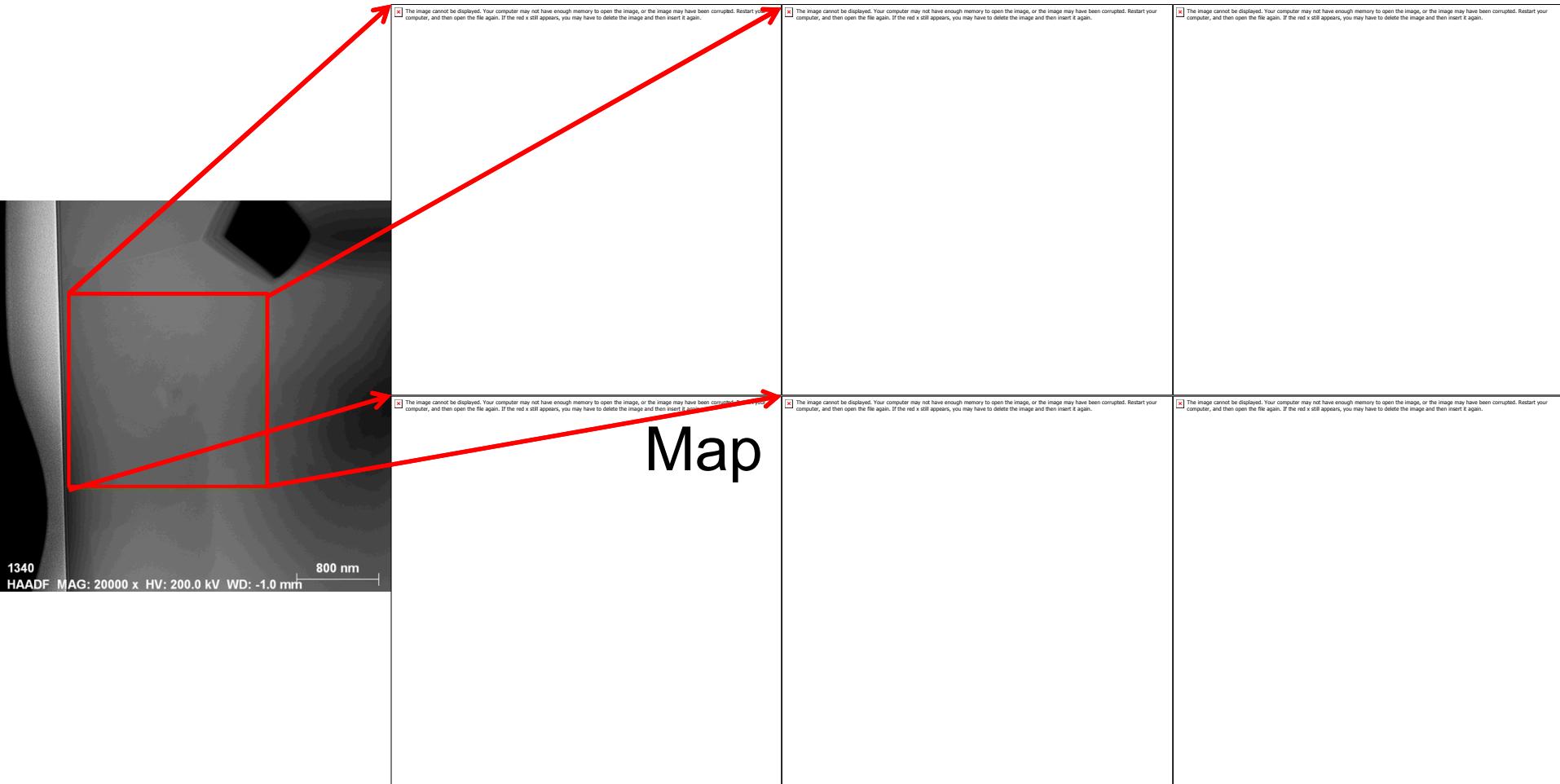
1309

MAG: 20000 x HV: 200.0 kV WD: -1.0 mm

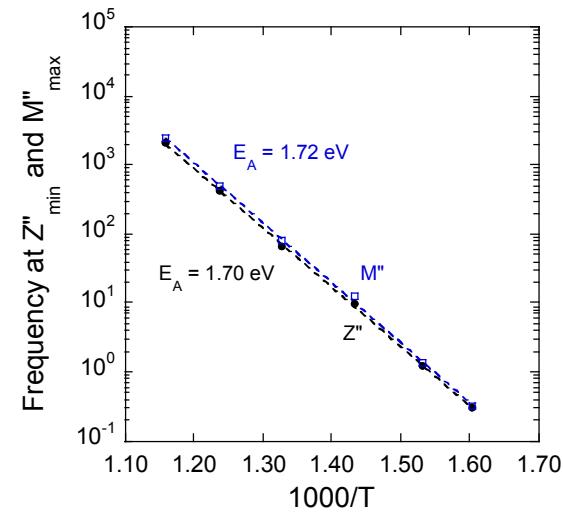
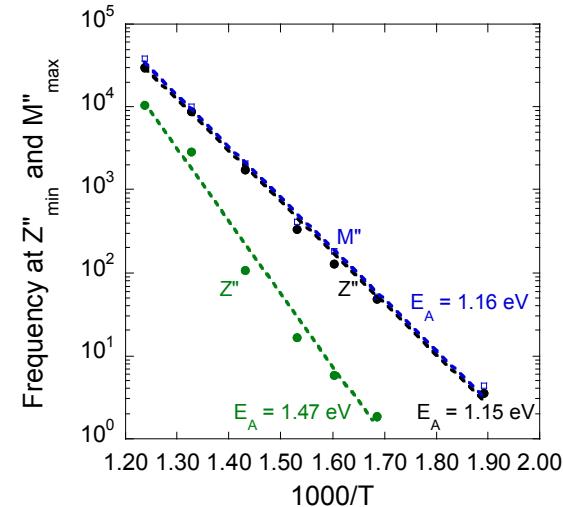
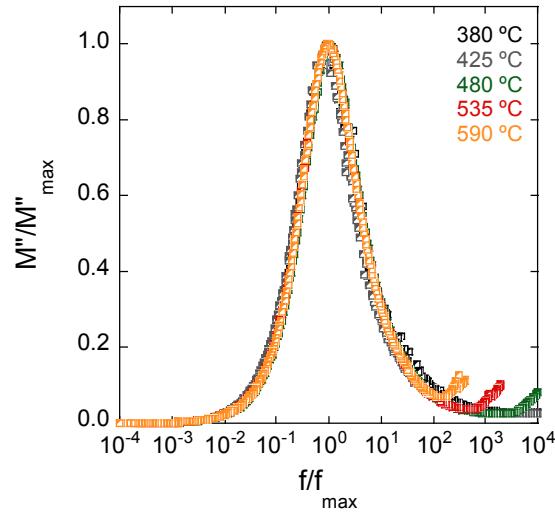
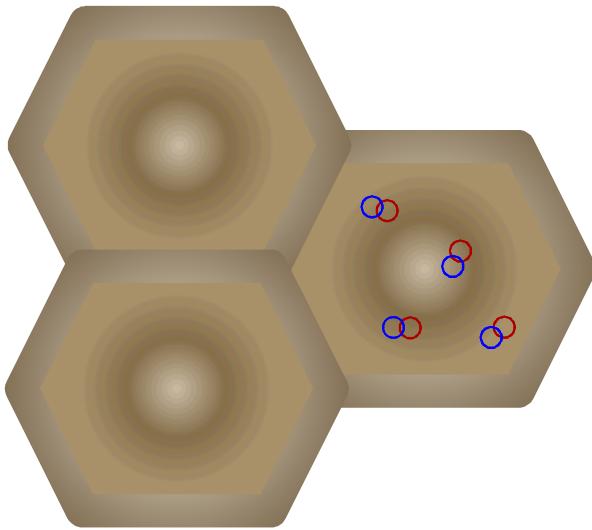
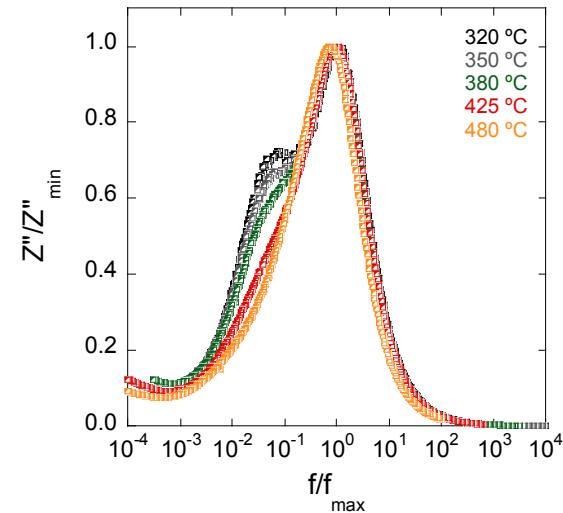
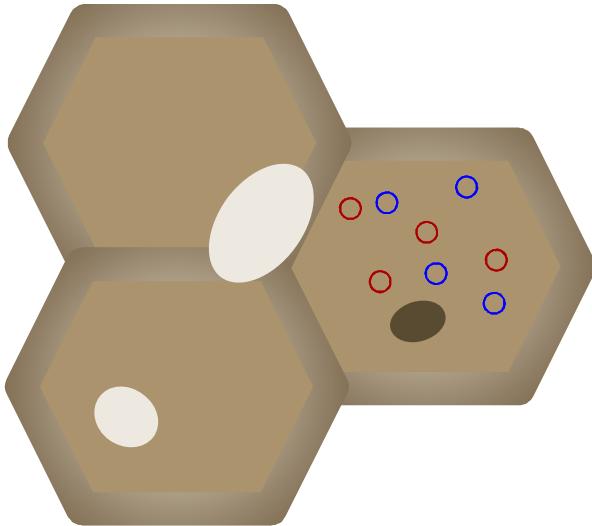
400 nm



Bi and Zn Co-segregation



Microscale Heterogeneity



with Prof. David Cann, Oregon State University