

Linearity for Ca^{2+} -Doped CeBr_3 Scintillating Materials

Paul Guss
Principal Investigator

Symposium on Radiation Measurements and Applications (SORMA XV)
June 9–13, 2014

This work was done by National Security Technologies, LLC, under Contract No. DE-AC52-06NA25946
with the U.S. Department of Energy and supported by the Site-Directed Research and Development Program.



Nevada National Security Site

Managed and Operated by National Security Technologies, LLC

Vision – Service – Partnership

Goals

- **Goal:** This project seeks to develop low-cost, high-resolution gamma scintillators based on new fracture-resistant alloys of cerium bromide. These alloys will eliminate limitations imposed by self-activity and fracture of currently available lanthanum-based scintillators.
- **Objectives:**
 - (1) Develop low-cost, high-resolution gamma scintillators based on new fracture-resistant alloys of CeBr_3 .
 - (2) Further improve crystals by eliminating radioisotope impurities in the raw materials to enable large, high-sensitivity gamma spectrometers for radioisotope identification and low count rate assay of SNM.
 - (3) Demonstrate significant performance advance for applications of importance to DOE missions.



Overview

- Spectroscopy characterization of the crystal delivered by Radiation Monitoring Devices (RMD) to the Remote Sensing Laboratory (RSL).
- Sandia National Laboratories (SNL) mapped out the solid solubility of Ca(II) in CeBr_3 .
- Pin Yang at SNL determined the material composition by using mass spectrometry techniques (inductively coupled plasma mass spectrometry [ICP-MS]). A comparison of SNL's measured compositions of the RMD fragments was made to assist in characterizing detector performance.
- Pin Yang and Patrick Doty, both of SNL, analyzed the calcium concentrations.



Background

Based on the recent literature on strengthening mechanisms (M. N. Sinha, P. S. Nicholson, “Effect of impurities on the strengthening of CaF_2 single crystals,” *J. Mater. Sci.* **12** (1977) 1451–1462), there are compelling requirements to research and share several ideas, questions, and answers. Sinha and Nicholson’s paper on aliovalent strengthening of CaF_2 attempts to determine mechanisms for low and high temperatures. The authors found that Y(III) gave an order of magnitude greater increase in critical resolve shear stress (CRSS) than Na(I). They conclude the long-range retarding force on dislocations at high temperature is likely due to the induced reorientation of Na(I)/F-vacancy or Y(III)/F-interstitial dipoles in the stress fields of moving dislocations (Snoek effect). This suggested role of the Snoek effect is in accord with analysis of the athermal regime in recent papers on Y-stabilized zirconia and other materials. It then occurs, if the authors are correct, that if anion interstitials balance the charge for the higher-valence cation, then the difference in strength may be related to the higher mobility of interstitials. Interstitial fluoride ion in CaF_2 may be more plausible than interstitial bromide; however, the CeBr_3 crystal structure does have large open channels. The SNL team will examine if the M(IV)/Br-interstitial seems like a feasible complex in CeBr_3 . Since the athermal (high-temperature) regime is probably more important during crystal growth, this work should explore if it makes sense to emphasize M(IV) cations.



SNL

- The amount of calcium in these CeBr_3 samples was determined by an induction coupled plasma mass spectrometer. These data will be used to extract the solubility limit.



Figure 1. Packaged scintillator of 1.9% Ca^{2+} -doped CeBr_3



Figure 2. Emission spectra

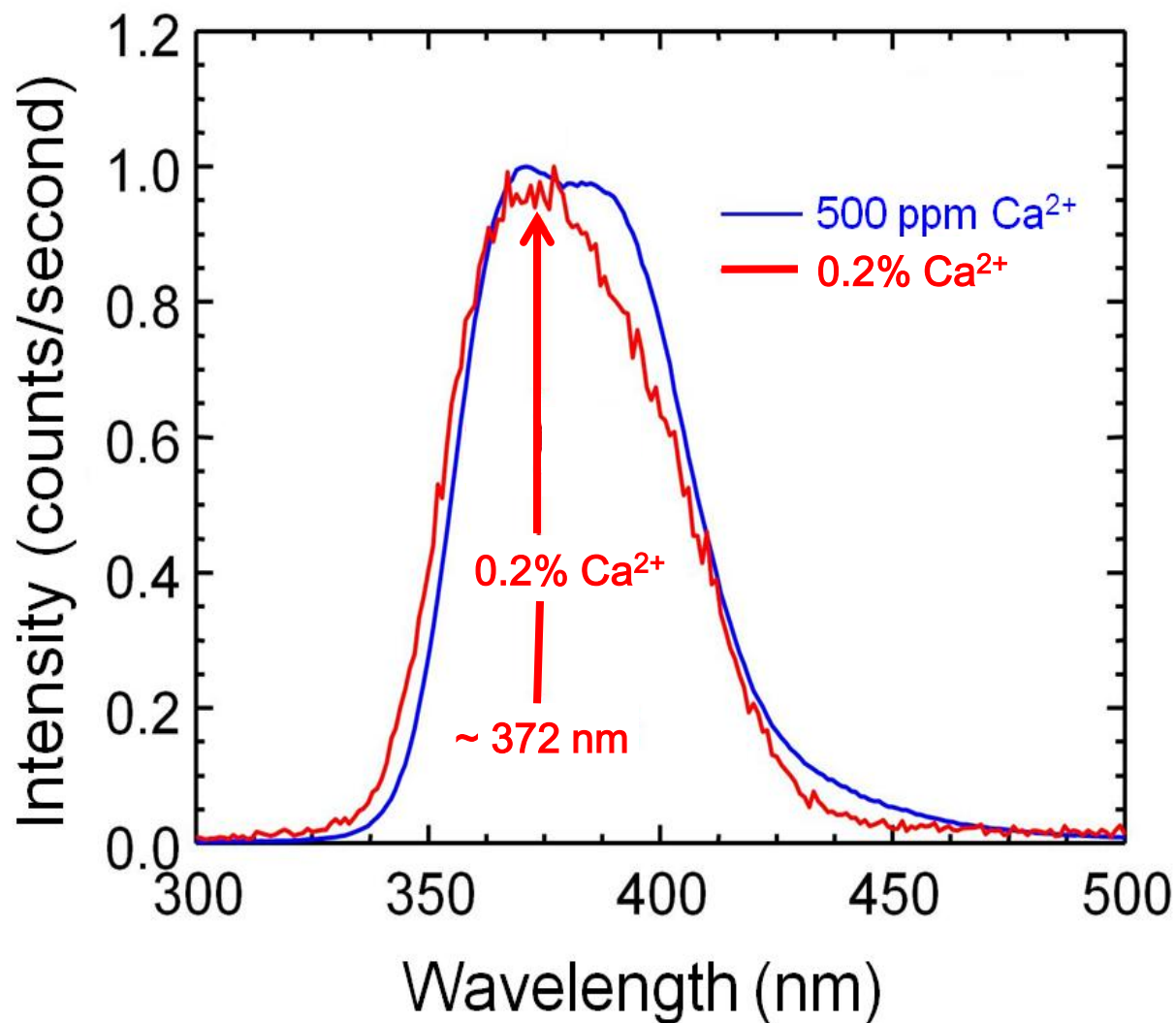


Figure 3. Relative light yield proportionality of Ca^{2+} -doped CeBr_3

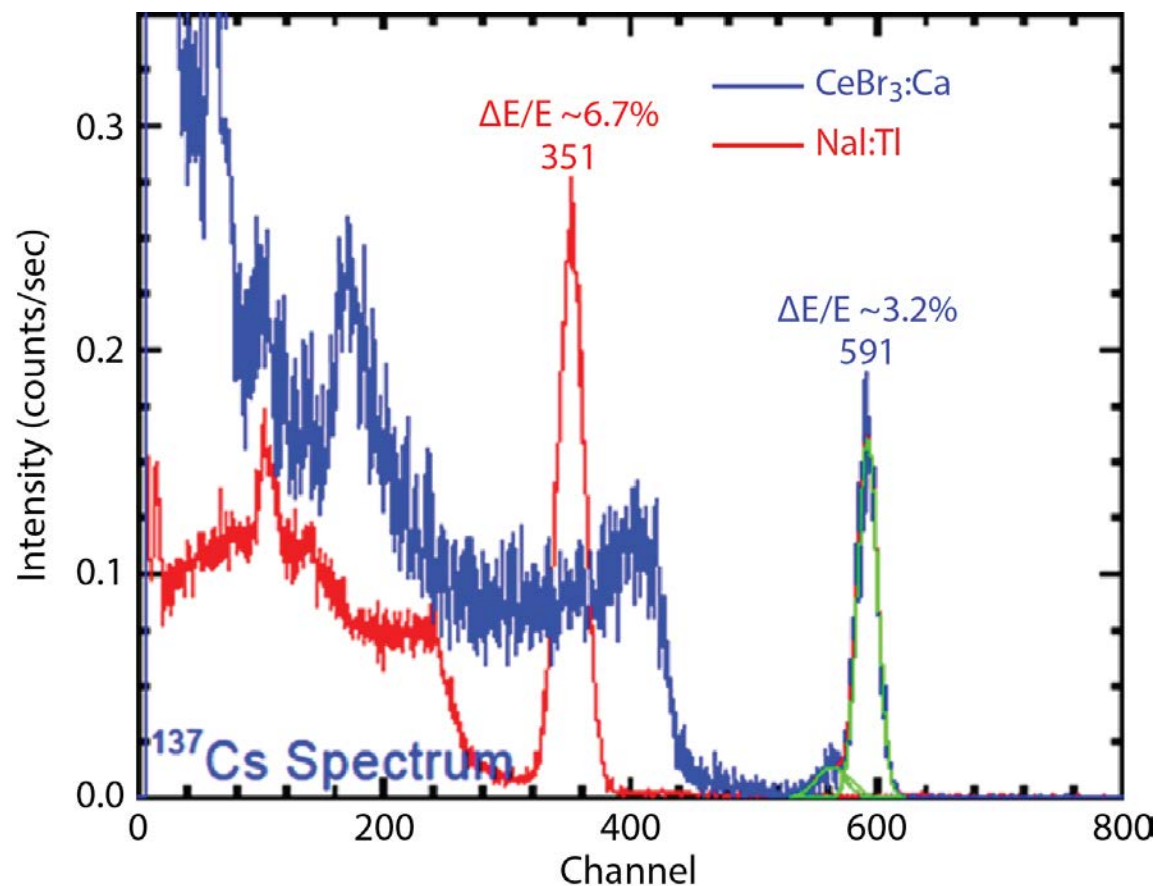


Figure 4. Relative light yield proportionality of Ca^{2+} -doped CeBr_3

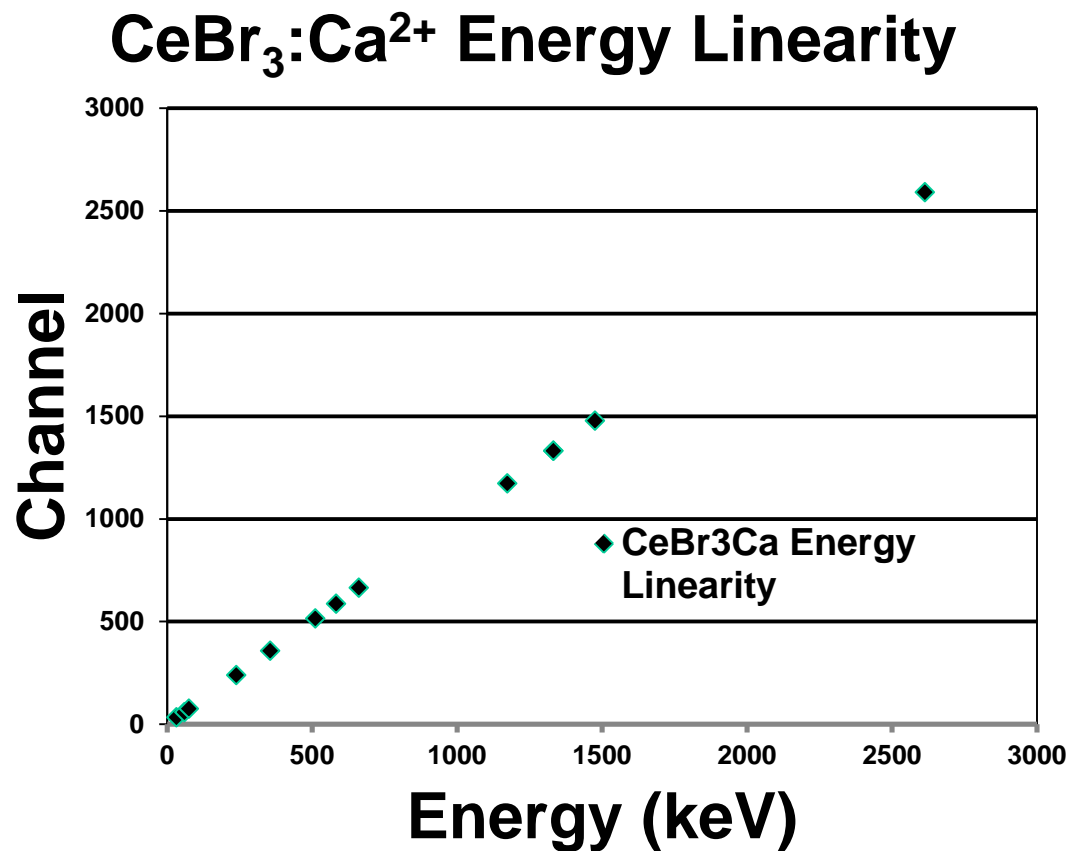


Figure 5. Energy resolution of Ca^{2+} -doped CeBr_3

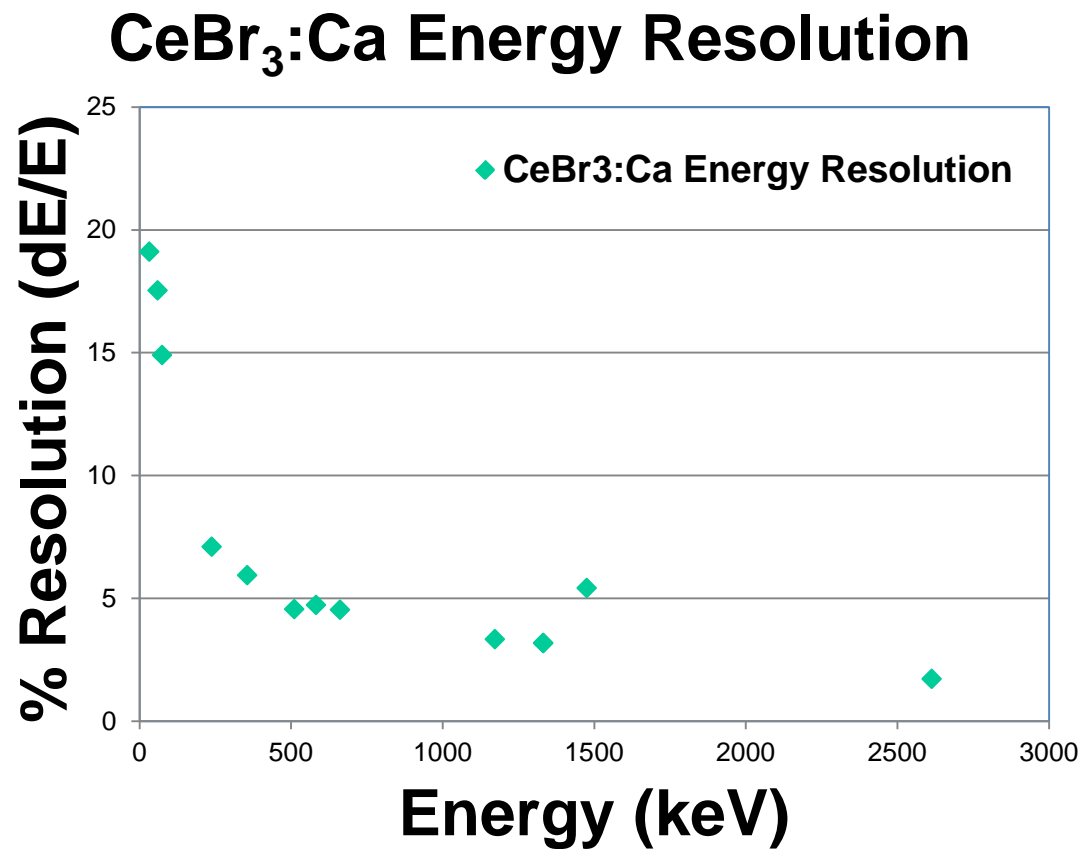


Figure 6. (a) 662 keV spectrum with standard CeBr_3 with ~4% FWHM, (b) 662 keV spectrum with $\text{CeBr}_3:\text{Ca}^{2+}$ with ~3.2% FWHM, (c) improved proportionality for $\text{CeBr}_3:\text{Ca}$ over standard CeBr_3

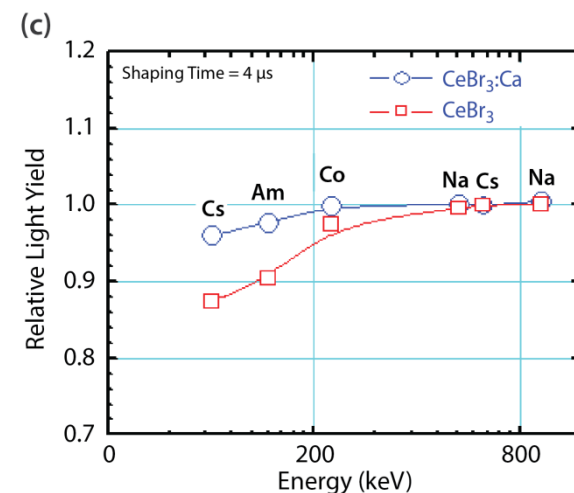
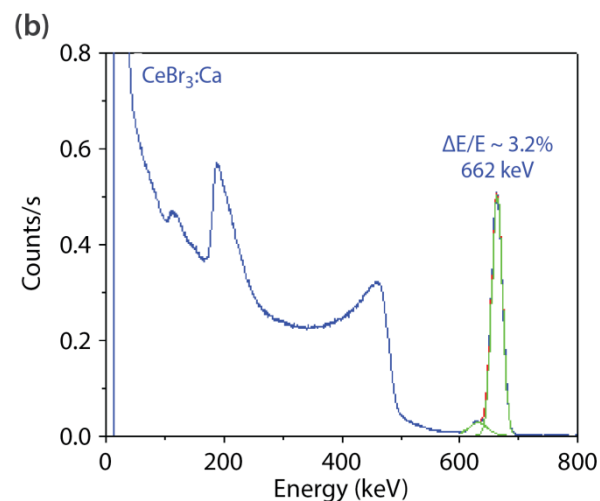
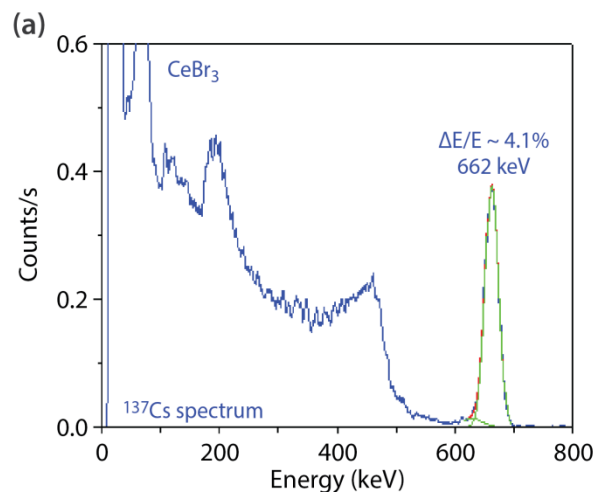


Figure 7. Pulse Shape

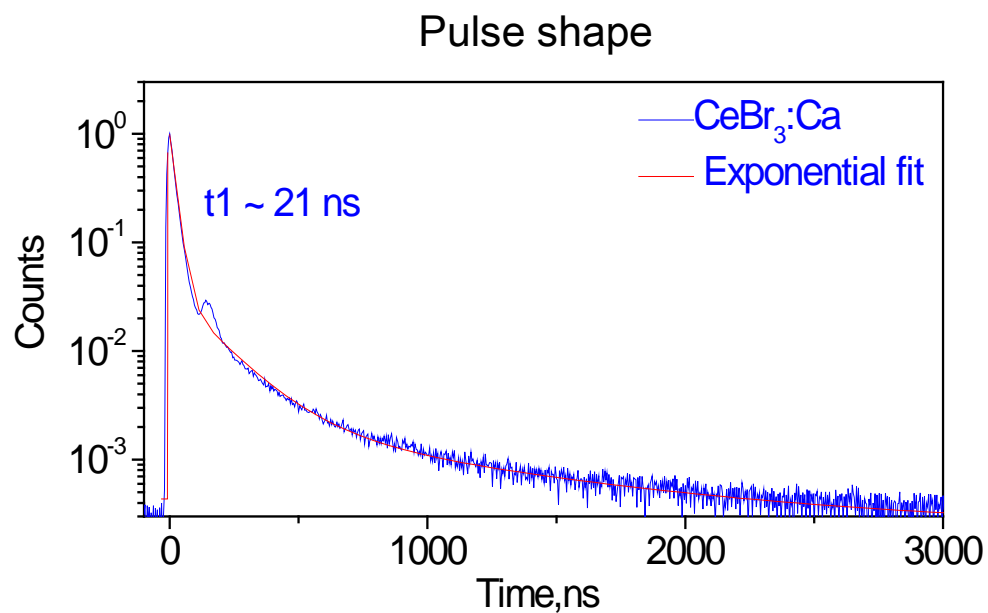


Figure 8. Comparison of measurements with the Ca^{2+} -doped CeBr_3 crystal in temporary test can (3.7%) versus the permanent can (4.0%)

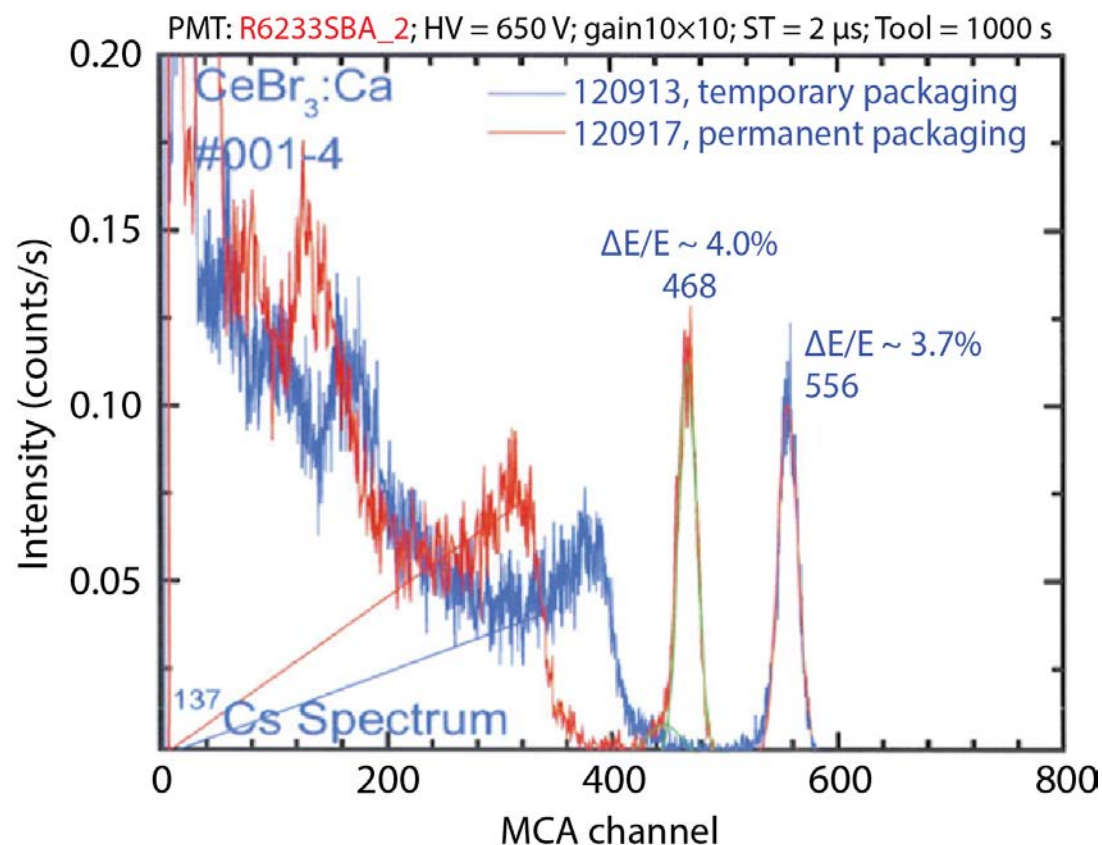


Figure 9. ^{137}Cs Spectrum with the Ca^{2+} -doped CeBr_3 crystal in the permanent can (4.5%)

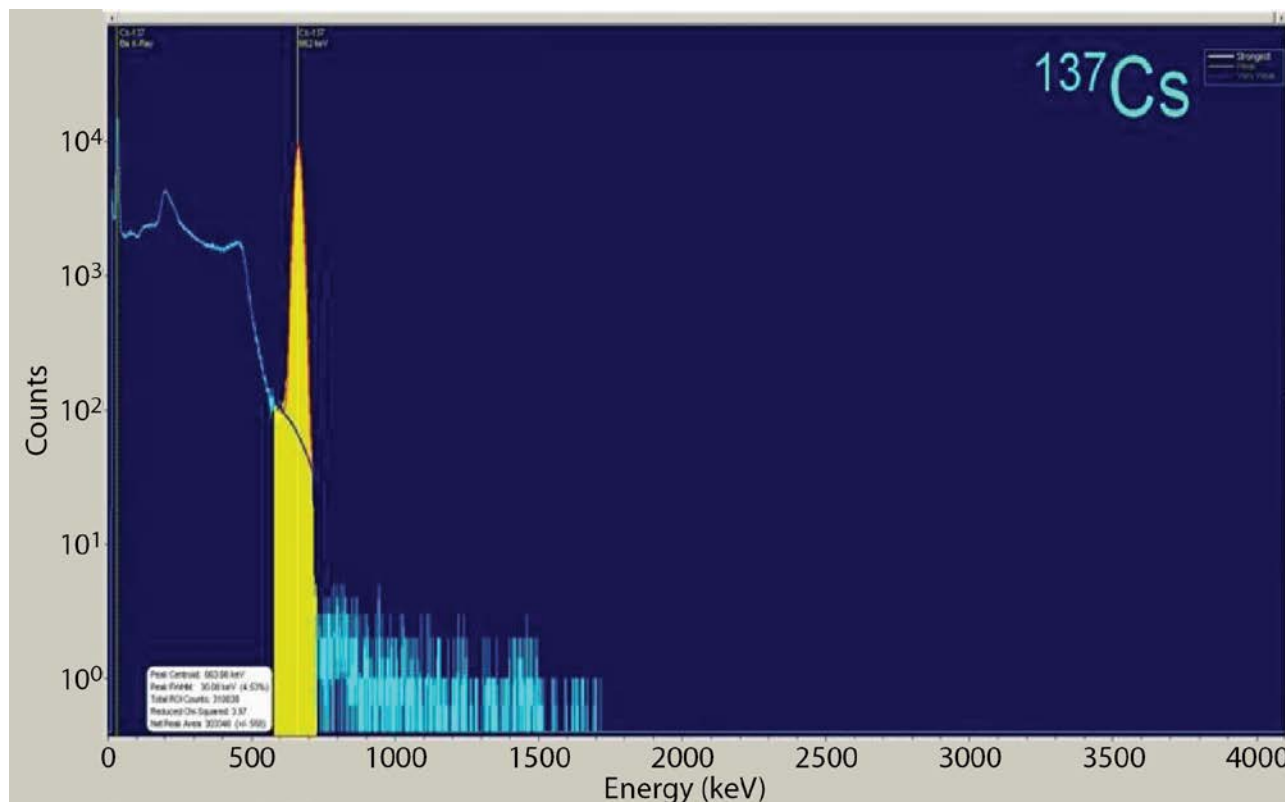


Figure 10. ^{241}Am , ^{60}Co , ^{228}Th , and background spectra with the Ca^{2+} -doped CeBr_3 crystal in the permanent can

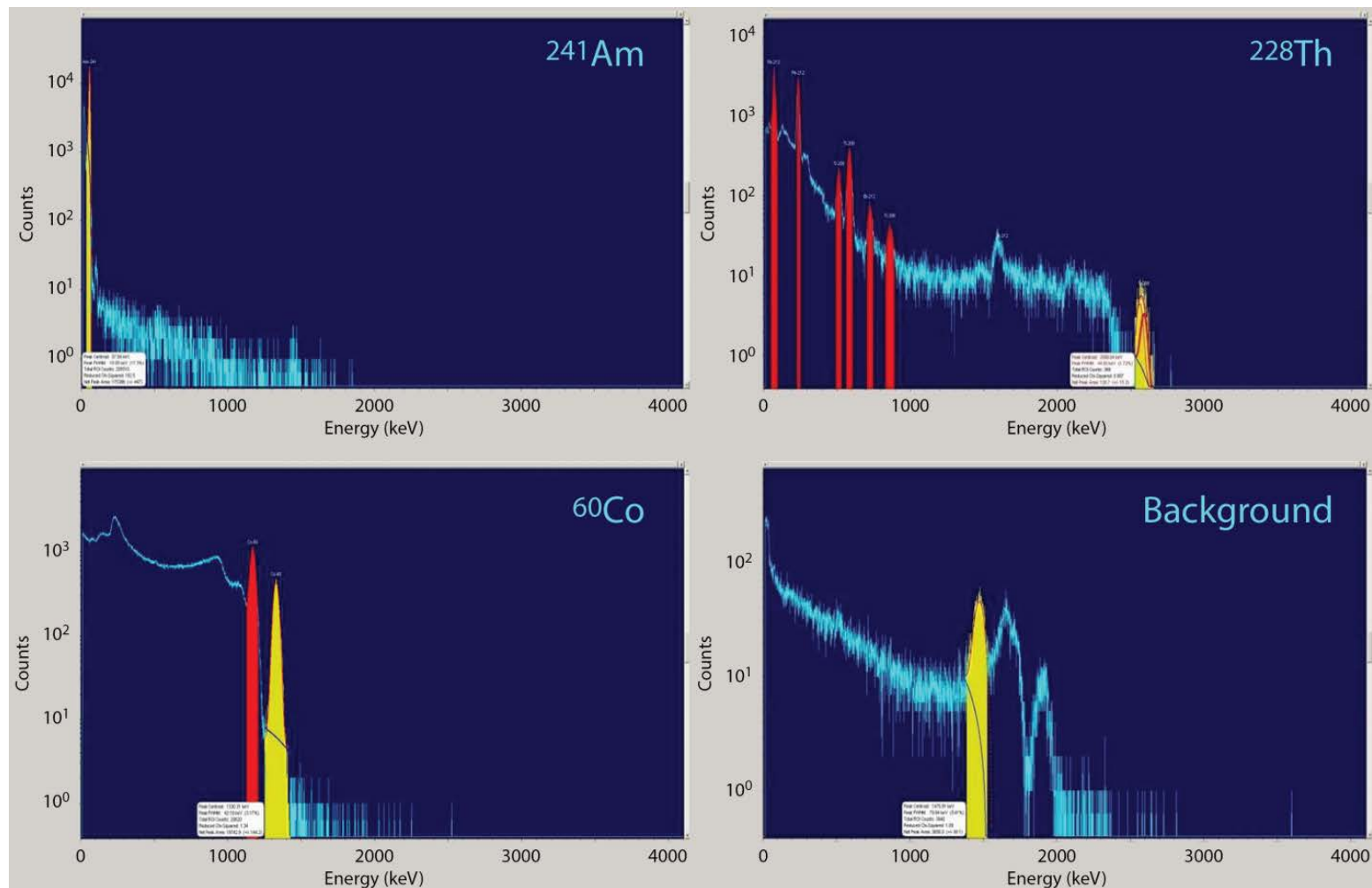


Table 1. Inductively coupled plasma mass spectrometry (ICP-MS) chemical analysis. ICP-MS assays for Ca^{2+} -doped CeBr_3 ; nominal composition 2 wt% CaBr_2 in CeBr_3

Sample	Calcium Concentration	95% Confidence Limit
#1	0.0238 wt%	0.0014 wt%
#2	0.0212 wt%	0.0015 wt%
#3	0.0192 wt%	0.0015 wt%

Average = 0.0214 ± 0.0102 wt.%



Evaluation of Atomic Percentage of Calcium

$$\begin{aligned}\text{Atomic \%} &= 40.078x / [(140.116 (1 - x) + 40.078x + 79.904 (3 - x))] \\ &= 0.00228 \\ &= 0.228\%\end{aligned}$$

Average calcium weight % concentration,
 $x = 0.000214$, in our complex of $\text{Ce}_{(1-x)}\text{Ca}_x\text{Br}_{(3-x)}$,

$$\text{Atomic \%} = 0.228 \text{ atomic\%}$$



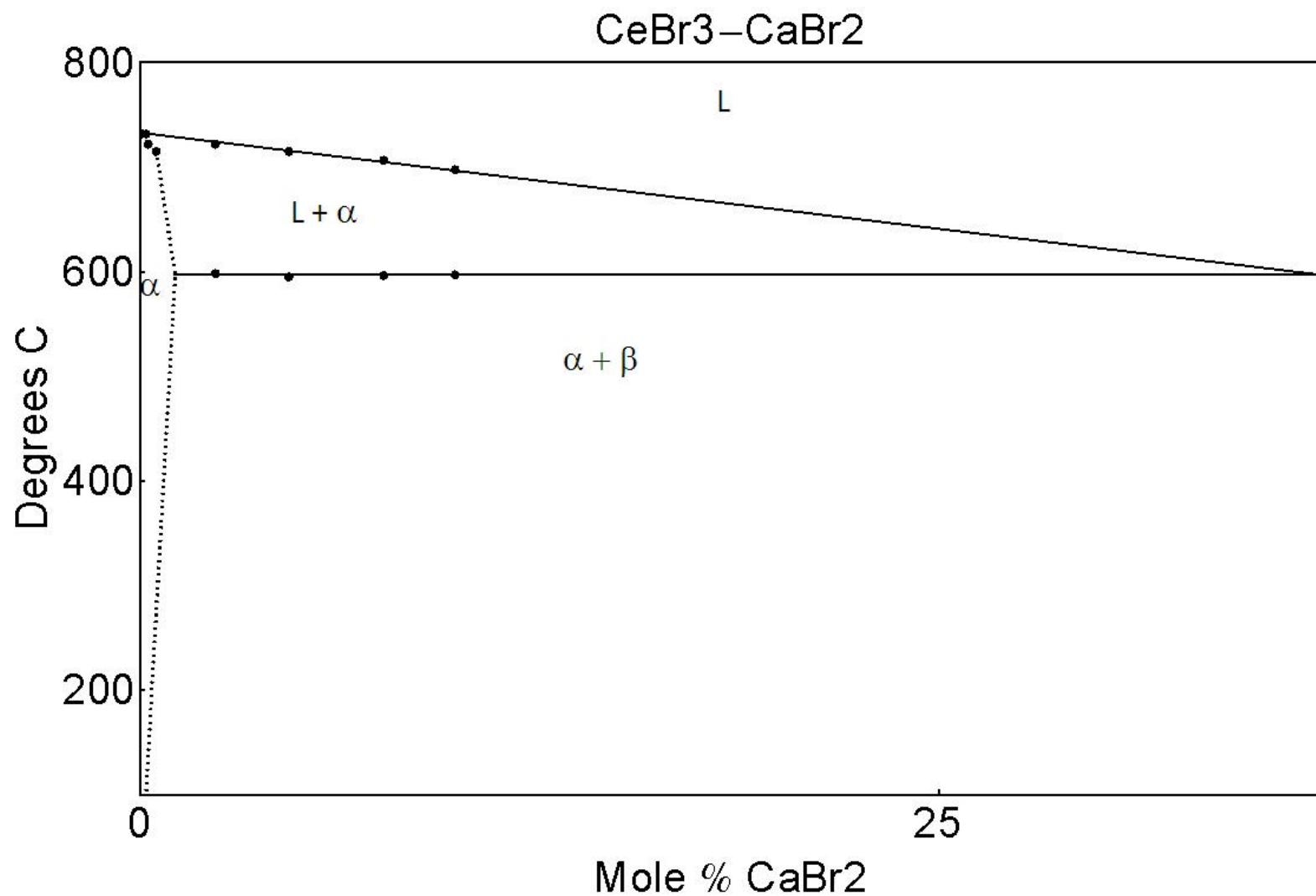
Table 2. Differential scanning calorimetric (DCS) results for CeBr_3 – CaBr_2 mixtures in the range 0–10 mole %

Mole % CaBr_2	T_m (°C)	T_e (°C)
0	715.3	—
0.2	732.1	—
2.35	721.8	598.3
4.65	715.2	595.4
7.63	706.8	596.2
9.85	697.3	597.0

The mixtures show linear trends, including an apparent eutectic temperature T_e near 597°C for calcium concentrations $\geq 2.35\%$.



Figure 11. Phase diagram for $\text{CeBr}_3\text{--CaBr}_2$



**Table 3. Measured total light yield data;
calculated and estimated total light yield**

Material	Peak FWHM (%)	Est. Total Light Yield (ph/MeV)	Reference
CeBr ₃	5.6	68,000	Guss
CeBr ₃ + 0.2% CaBr ₂	3.2	62,000	Guss
CeBr ₃ + 500 ppm CaBr ₂	4.5	65,200	RMD 2009
BGO	10	8,200	Harrison
CeBr ₃ + 500 ppm SrBr ₂	4.9	48,600	Harrison
CeBr ₃ + 500 ppm CdBr ₂	6.7	34,800	Harrison
CeBr ₃ + 500 ppm ZrBr ₄	6.6	37,600	Harrison
CeBr ₃ + 1000 ppm HfBr ₄	5	52,900	Harrison
LaBr ₃ + 5% CeBr ₃	4.2	85,100	Harrison
NaI:Tl	8.3	36,700	Harrison



Technical Issues

- Need to determine concentration of calcium. If it cracks at saturation, may not determine accurate concentration.
- Need to benchmark below solubility limit.
- Need to examine other dopants, such as Ba, Sr, and Cd (SDRD project convincingly eliminated Pb).



Future Direction

- Complete analysis of spectroscopy benchmarking
- Calculate with a less dilute doping (i.e., something less than 2%) of Ca in CeBr_3 . This will help with the assessment what doping concentration will lead to changes in optical and mechanical properties.
- Perform hardness measurements to see if the aliovalent approach hardened the crystal as expected
- Other reasonable research areas include:
 - Calculate density functional theory (DFT) model with less-dilute doping
 - Perform growth with lower concentrations of calcium
 - Perform similar growth runs using strontium



Summary

- New DFT simulations demonstrate a capability for predicting properties of doped CeBr_3 materials that is unavailable elsewhere but is critically needed to study the property-limiting valence phenomena in ionic compounds
- Assessed concentrations and the solubility limit
- RSL benchmarked the Ca^{2+} -doped CeBr_3 crystal
- The Ca^{2+} -doped CeBr_3 crystal has improved energy resolution (i.e., 3.2%)
- The Ca^{2+} -doped CeBr_3 crystal has improved linearity over the pure CeBr_3 crystal



Acknowledgments

Bryan Wong (SNL) performed the DFT calculations.

Patrick Doty (SNL) provided advice and guidance.

RMD prepared the crystal and performed some of the characterization.

Marlene Bencomo (UNM) assisted with IC-PMS analysis.

The author also thanks Thomas Stampahar, Michael Lukens, John O'Donoghue, Michael Mohar, and Sanjoy Mukhopadhyay of NSTec for their contributions to this work.



References

- [1] Shah, CeBr₃ Scintillators for Gamma-Ray Spectroscopy DOE Contract No. DE-AC03-76SF00098, RMD.
- [2] Guss, NIM A **608**, 2 (2009) 297.
- [3] Guss, SPIE **7805** (2010) L-1.
- [4] Shah, IEEE Trans. Nucl. Sci., **51**, no. 5 (2004) 2395–2399.
- [5] Doty, Proc. Mat. Res. Soc. **1038** (2008) 1-8.
- [6] Harrison, SPIE **7806** (2010) M-1.
- [7] Pletka, Physica Status Solidi **39** (1977) 301–311.
- [8] Setyawan, Wahyu; Curtarolo, Stefano. "High-throughput electronic band structure calculations: Challenges and tools." Comp. Mat. Sci. **49**: (2010) 299–312.
- [9] Li, Chenliang, Biao Wang, Rui Wang, Hai Wang. "First-principles studies on the electronic and optical properties of CeCl₃ and CeBr₃." Solid State Communications 144 (2007) 220–224.
- [10] Periodic Table of the Elements: LANL. <http://periodic.lanl.gov/list.shtml> (accessed February 6, 2012).
- [11] Snoek, J., "Effect of small quantities of carbon and nitrogen on the elastic and plastic properties of iron," *Physica* **8**, 7 (1941) 711–733.
- [12] Sinha, M. N., P. S. Nicholson, "Effect of impurities on the strengthening of CaF₂ single crystals," *J. Mater. Sci.* **12**, 7 (1977) 1451–1462.
- [13] Hall, R. N., "Variation of the distribution coefficient and solid solubility with temperature," *J. Phys. Chem. Solids* **3**, 1–2 (1957) 63–73.
- [14] Fischler, S., "Correlation between maximum solid solubility and distribution coefficient for impurities in Ge and Si," *J. Appl. Phys.* **33**, 4 (1962) 1615.
- [15] P. Guss, et al., *Site-Directed Research and Development*, FY 2013, National Security Technologies, LLC, Las Vegas, Nevada, 2014.

