

LA-UR-

Approved for public release;
distribution is unlimited.

12-00170

Title: Theoretical study of electrochemical processes on platinum nanostructures

Author(s): Matanovic Ivana, 240672
Garzon Fernando, 102961
Henson Neil, 119604

Intended for: PGM catalyst project meeting at University of New Mexico, Albuquerque, New Mexico, USA. January 18th 2012



Los Alamos National Laboratory, an affirmative action/equal opportunity employer, is operated by the Los Alamos National Security, LLC for the National Nuclear Security Administration of the U.S. Department of Energy under contract DE-AC52-06NA25396. By acceptance of this article, the publisher recognizes that the U.S. Government retains a nonexclusive, royalty-free license to publish or reproduce the published form of this contribution, or to allow others to do so, for U.S. Government purposes. Los Alamos National Laboratory requests that the publisher identify this article as work performed under the auspices of the U.S. Department of Energy. Los Alamos National Laboratory strongly supports academic freedom and a researcher's right to publish; as an institution, however, the Laboratory does not endorse the viewpoint of a publication or guarantee its technical correctness.

Theoretical study of electrochemical processes on platinum nanostructures

Ivana Matanovic^a, Fernando Garzon^b and Neil Henson^a

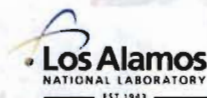
^a Los Alamos National Laboratory, Physics and Chemistry of Materials Group, New Mexico 87545, USA

^b Los Alamos National Laboratory, Electronic and Electrochemical Materials and Devices Group, New Mexico 87545, USA

Future energy security concerns demand a transition from fossil-fuels to a more environmental benign energy sources. The use of fuel cells to directly convert chemical energy of fuels to electricity is a promising route for achieving this. Efficient fuel cell performance however requires the engineering of platinum group metal catalysts with higher mass activity and more stability. Along that line, we have undertaken a comprehensive and systematic study of the structure, reactivity and stability of different single and multi-walled platinum nanotubes using the plane wave, pseudopotential implementation of DFT. The change in the catalytic activity of the nanomaterial induced by the structural changes (size and the chirality) is studied by calculating equilibrium adsorption potentials for oxygen reduction reaction (ORR) intermediates and by constructing free energy diagrams in the ORR dissociative mechanism network. In addition, the stability of the platinum nanotubes both in gas and aqueous environment is investigated in the terms of relative electrochemical dissolution shifts and by determining the most stable state of the material as a function of pH and potential as represented in Pourbaix diagrams.

Theoretical study of electrochemical processes on platinum nanostructures

Ivana Matanović, Fernando Garzon, Neil Henson
Physics and Chemistry of Materials
Los Alamos National Laboratory, Los Alamos, USA



UNCLASSIFIED



Slide 1

Operated by Los Alamos National Security, LLC for the U.S. Department of Energy's NNSA



Motivation

Reducing the ORR overpotential / cost:

(1) **alloying** platinum with platinum group metals

(2) **nanostructures**: nanotubes and nanoparticles

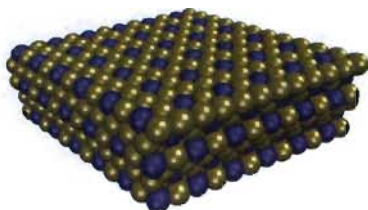


Figure: Pt₃Ni(111) surface

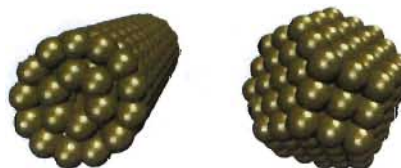


Figure: (6,6)@(13,13) MWPtNT nanotube and 2nm Pt₂₀₁ cluster



UNCLASSIFIED

Slide 2

Operated by Los Alamos National Security, LLC for the U.S. Department of Energy's NNSA



Motivation



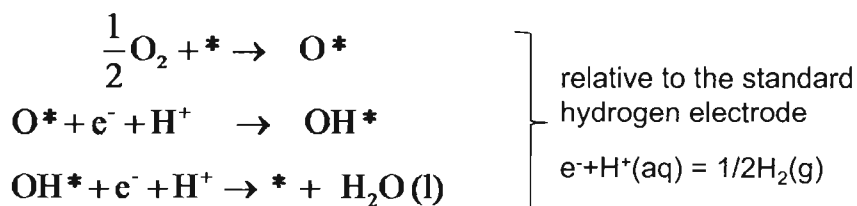
study the influence of alloying component **concentration and distribution** on the **ORR activity** and **stability** in aqueous environment



study the effect of **size and structure** of a nanomaterial on the ORR activity and stability in aqueous environment

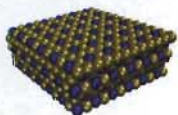
Methodology - study of ORR mechanism

Reactions connecting different states of the metal surface(*) in the ORR mechanism



Free gibbs energy of the reactions (Norskov et al. J. Phys. Chem. B 2006, 110, 21833)

$$\begin{aligned}
 \Delta G_{\text{w,water}} &= \Delta E_{\text{w,water}} + \Delta \text{ZPE} + T\Delta S \\
 \Delta G(U, \text{pH}, T = 298 \text{ K}) &= \Delta G_{\text{w,water}} \underbrace{- eU}_{\text{bias effect}} + \underbrace{kT \ln(10) \text{pH}}_{\text{correction for the free energy of H}^+}
 \end{aligned}$$



Oxygen Reduction Reaction on Pt-Ni alloys

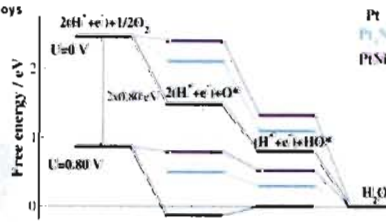
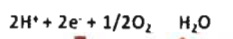
Conclusions

- catalytic activity – modification in the electronic structure induced by the specific subsurface composition
- the ORR overpotential was found to decrease



confirmed experimentally, ECS 220th meeting, Boston, NIST group

Oxygen reduction reaction on Pt-Ni alloys



Conclusions

shifts in the electrochemical dissolution potentials relative to Pt:

PtNi

- the least susceptible to corrosion

Pt₃Ni

- most susceptible to electrochemical dissolution of Pt monolayer
- most susceptible to poisoning of the surface by the formation of nickel oxide

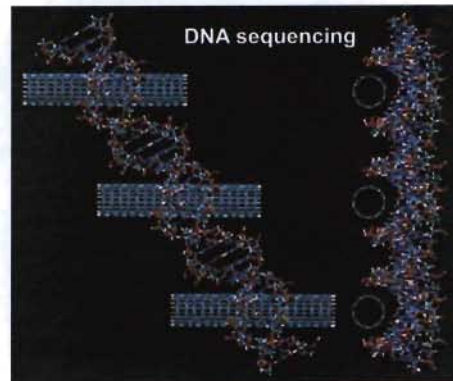
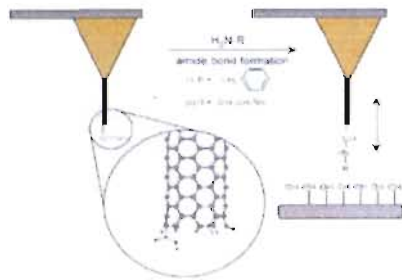
[dx.doi.org/10.1021/jp111930w](https://doi.org/10.1021/jp111930w) | *J. Phys. Chem. C* 2011, 115, 10640–10650



Oxygen Reduction Reaction on Pt nanotubes

carbon nanotubes, 90'

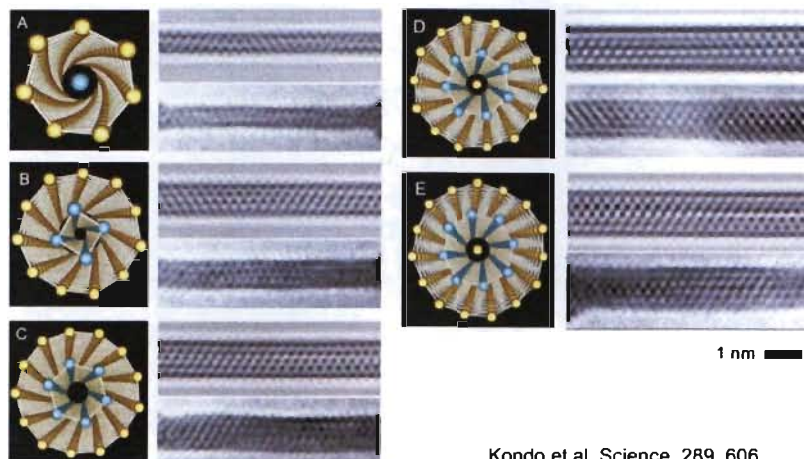
AFM probe tips



(bio)sensing, imaging, nanoelectronics

metal nanotubes, 2000'

- magic "structure" and conductance - nanoelectronics
- Au, Ag nanotubes less than 2nm thickness have been synthesized by an electron-beam technique in an UHV-TEM



Kondo et al. Science, 289, 606

metal nanotubes, 2000'

- Pt nanotubes – 1.0 and 0.5 nm in diameter

0.5 nm – six atomic rows coiling around tubes axes

1 nm – 13-6 multishell structure

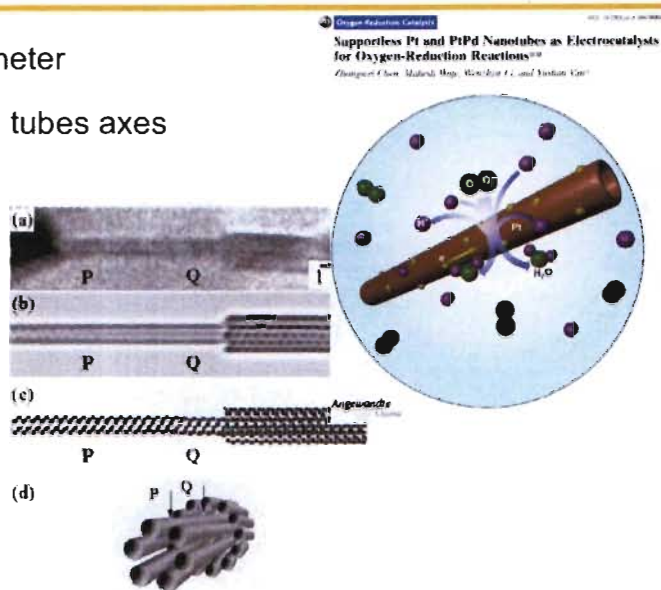
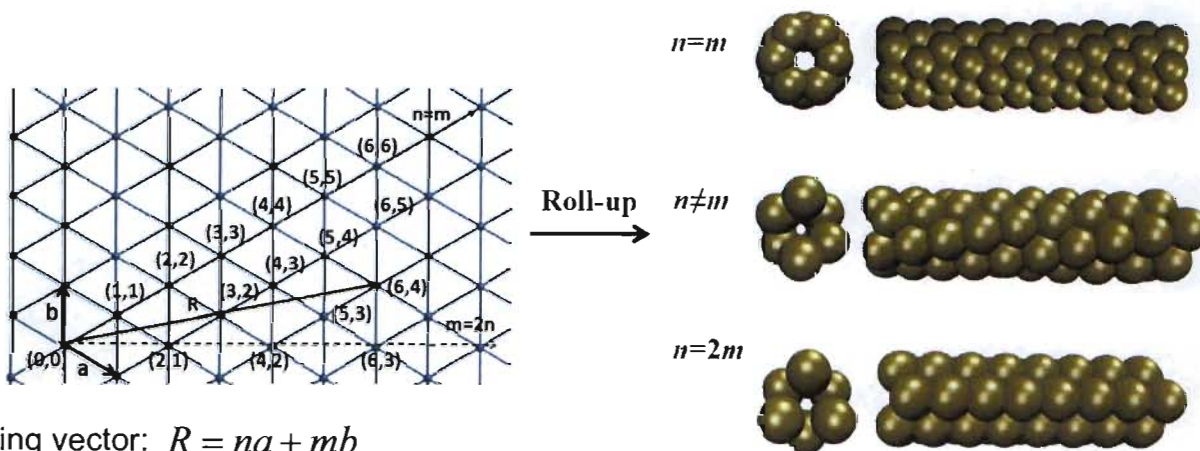


Figure: HR-TEM image of platinum nanotube
Y. Oshima et. al, Phys. Rev. B, 65, 121401 (2002)

Pt nanotubes

PtNT: Rolling-up Pt(111) sheet to form a tube

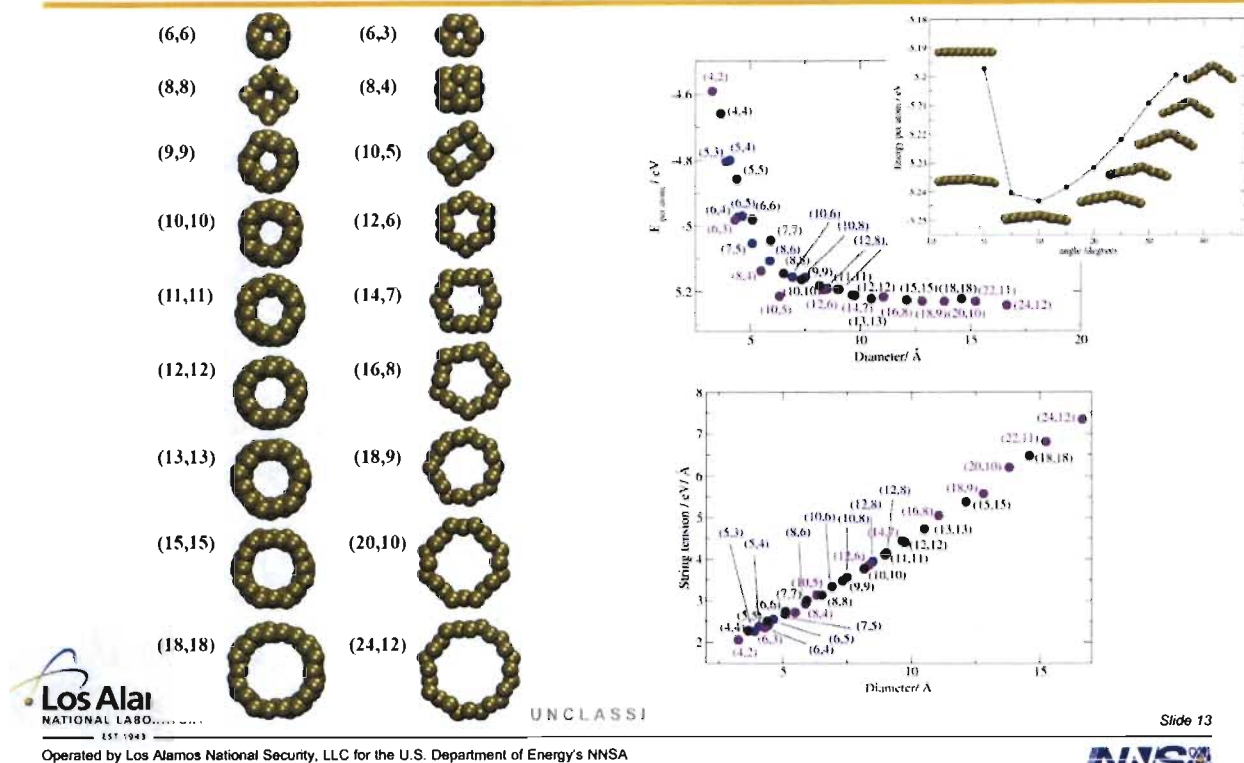


rolling vector: $R = na + mb$

$$r = \frac{\sqrt{2}a_c}{4\pi} \sqrt{n^2 + m^2 - nm}$$

$$a_c = 3.70 - 3.85 \text{ \AA}$$

Pt nanotubes



Pt nanotubes + oxygen

Table. Adsorption energies in eV and a shift in equilibrium adsorption potential in V for oxygen on the fcc site for different nanotubes and coverages

		0.25 ML		0.33 ML		0.5 ML	
		E_{ad}	ΔU_f	E_{ad}	ΔU_f	E_{ad}	ΔU_f
	Pt	-4.42	0.00	-4.25	0.00	-4.07	0.00
≈ 0.5 nm	(6,3)	-4.72	-0.15	-4.52	-0.14	-4.50	-0.22
	(6,4)	-4.70	-0.14	-4.31	-0.03	-4.44	-0.19
	(6,6)	-4.72	-0.15	-4.41	-0.08	-4.46	-0.20
≈ 1 nm	(12,6)	-4.29	+0.07	-3.98	+0.14	-3.96	+0.06
	(12,8)	-4.29	+0.07	-4.03	+0.11	-4.00	+0.04
	(12,12)	-4.18	+0.12	-3.94	+0.16	-3.97	+0.05
	(13,13)	-4.14	+0.14	-3.90	+0.17	-3.92	+0.08
	(6,6)@(13,13)	-4.05	+0.19	-4.21	+0.02	-4.06	+0.05

Los Alamos
NATIONAL LABORATORY
EST. 1943

UNCLASSIFIED

Slide 14

NNSA

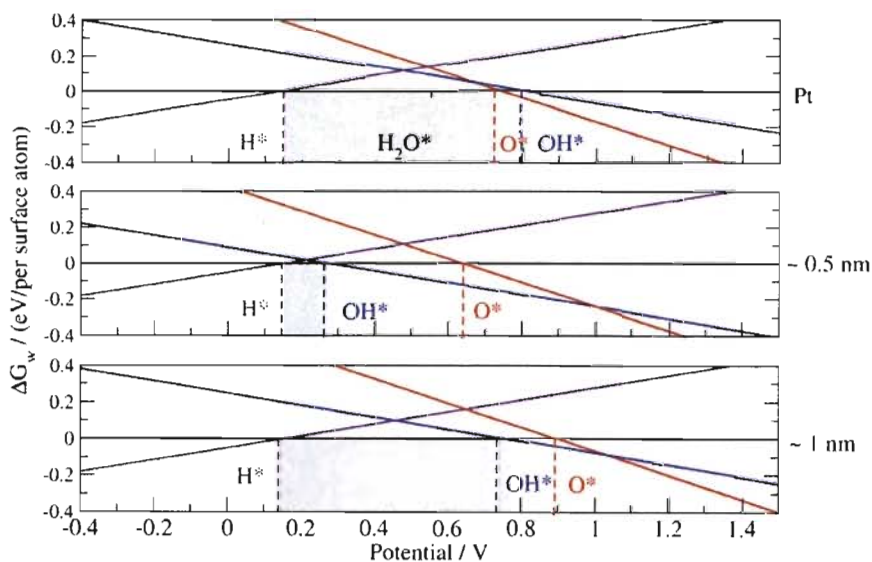
Pt nanotubes + hydroxyl

Table. Adsorption energies in eV and a shift in equilibrium adsorption potential in V for hydroxyl on the atop site for different nanotubes and coverages

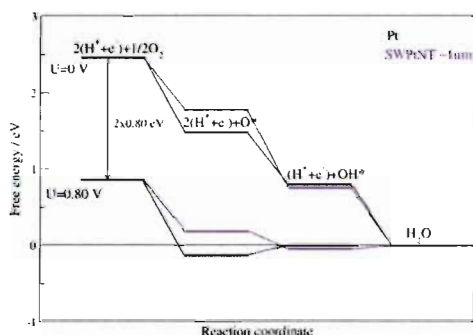
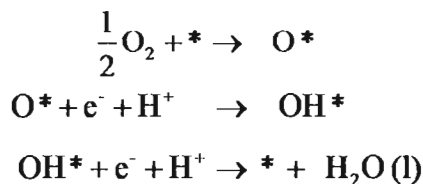
		0.25 ML		0.33 ML		0.5 ML	
		E_{ad}	ΔU_f	E_{ad}	ΔU_f	E_{ad}	ΔU_f
	Pt	-2.88	0.00	-2.92	0.00	-3.09	0.00
$\approx 0.5\text{nm}$	(6,3)	-3.70	-0.82	-3.47	-0.55	-3.76	-0.67
	(6,4)	-3.60	-0.72	-3.35	-0.43	-3.69	-0.60
	(6,6)	-3.54	-0.66	-3.54	-0.62	-3.52	-0.43
$\approx 1\text{nm}$	(12,6)	-2.92	-0.04	-2.98	-0.06	-2.73	+0.36
	(12,8)	-2.89	-0.01	-2.98	-0.06	-2.73	+0.36
	(12,12)	-2.89	-0.01	-2.97	-0.05	-2.91	+0.18
	(13,13)	-2.83	+0.03	-2.87	+0.03	-2.85	+0.24
	(6,6)@(13,13)	-3.07	-0.19	-3.13	-0.21	-3.20	-0.11



Pt nanotubes phase diagram



Dissociative oxygen reduction reaction (ORR) mechanism



PtNTs, $d \sim 1\text{nm}$

smaller ORR overpotential than
Pt(111), up to 100 meV

Figure: Free-energy diagrams for ORR over Pt(111) surfaces and SWPtNT for cell potentials $U=0.80\text{ V}$

Dissociative oxygen reduction reaction (ORR) mechanism

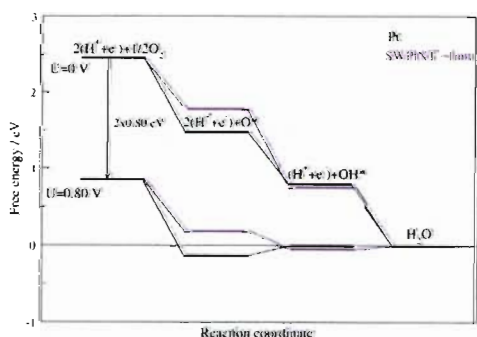
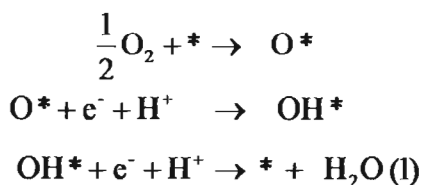
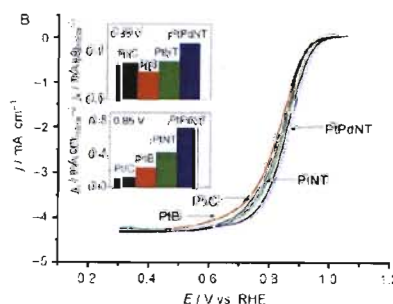


Figure: Free-energy diagrams for ORR over Pt(111) surfaces and SWPtNT for cell potentials $U=0.80\text{ V}$

In need for measurements – ORR curves



Chen et al, Angew. Chem. Ind. Ed. 2007, 46, 4060

Stability of the surfaces

What about the stability of these nanostructures?

Electrochemical dissolution can severely decrease the performance of the material

Stability of the surfaces

Estimate of the shift in the electrochemical dissolution potential

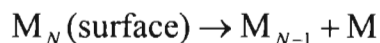


Table: surface cohesive energy of different tubes and the shift in the electrochemical dissolution potential relative to Pt(111)

reaction	$M_N(\text{tube}) \rightarrow M_{N-1} + M$	
	$\Delta E/\text{eV}$	$\Delta U_{\text{eq}}/\text{V}$
Pt(111)	6.55	0.00
(6,3)	4.11	-1.22
(6,4)	5.17	-0.70
(6,6)	5.20	-0.68
(12,6)	5.53	-0.51
(12,8)	5.52	-0.52
(12,12)	6.05	-0.25
(13,13)	6.29	-0.13
(6,6)@(13,13)	5.98	-0.27

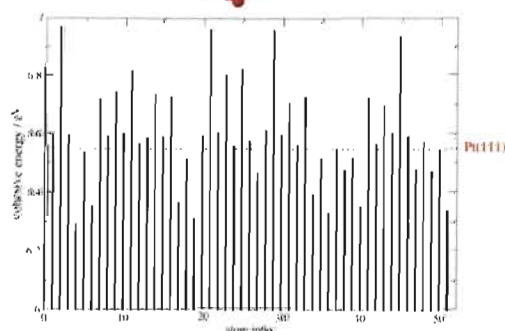


Figure. Surface cohesive energy of atoms in (13,13) tube

Stability of the surfaces

Estimate of the shift in the electrochemical dissolution potential

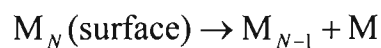


Table: surface cohesive energy of different tubes and the shift in the electrochemical dissolution potential relative to Pt(111)

reaction	$M_N(\text{tube}) \rightarrow M_{N-1} + M$	
	$\Delta E/\text{eV}$	$\Delta U_{\text{corr}}/\text{V}$
Pt(111)	6.55	0.00
(6,3)	4.11	-1.22
(6,4)	5.17	-0.70
(6,6)	5.20	-0.68
(12,6)	5.53	-0.51
(12,8)	5.52	-0.52
(12,12)	6.05	-0.25
(13,13)	6.29	-0.13
(6,6)@(13,13)	5.98	-0.27

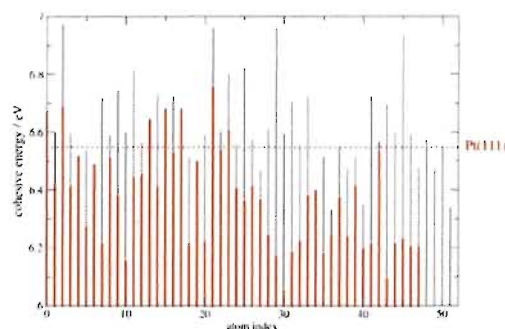


Figure. Surface cohesive energy of atoms in (12,12) tube

Pourbaix diagrams

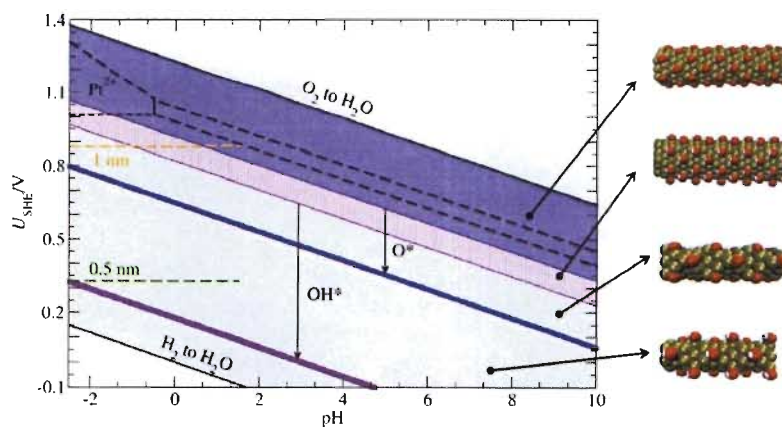
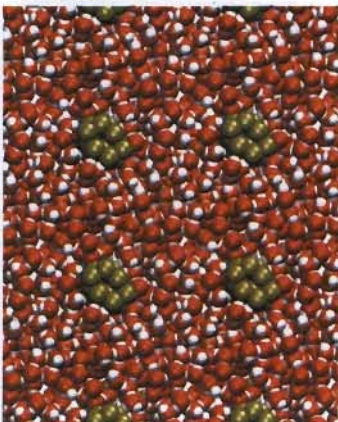


Figure: Calculated surface Pourbaix diagrams for Pt nanotubes compared to a bulk Pourbaix diagrams (black dashed lines)

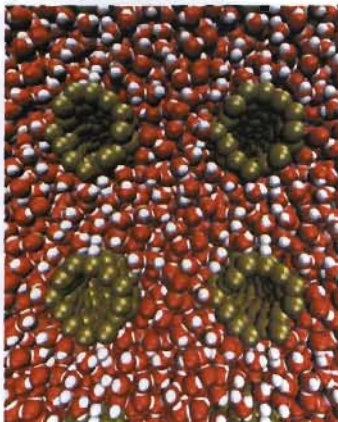
Pt nanotubes – ab initio MD simulations in water

- Aim** (1) characterize change of atomic and electronic structure on solvation
(2) structure of water around curved surfaces - water-surface interface models

(6,6) and

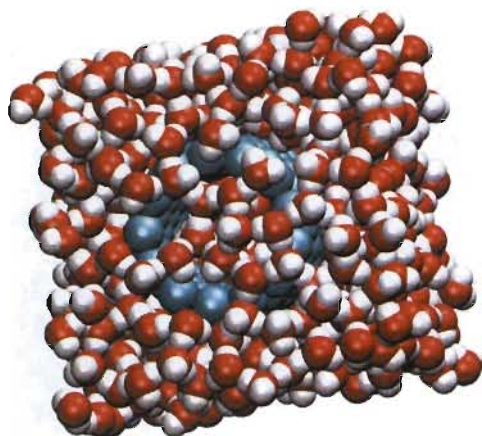


(13,13) SWPtNT in water



~800 atom cell, 1300 MD steps
in 24h, 480 processors, average
~1min/step

Pt nanotubes – ab initio MD simulations in water



20 Å length tube, $d=10.9$ Å in a 30 Å length
simulation box

bigger tubes accommodate water and ORR
intermediates

Conclusions

- smaller nanotubes (~0.5 nm) bind oxygen/hydroxyl more strongly than Pt(111)

larger nanotubes (~1 nm) bind oxygen/hydroxyl comparable or weaker than Pt(111)

reduced ORR overpotential – SWPtNT with a diameter > 1nm

- control size/chirality – fine tuning of reactivity → separation of metal nanotubes by geometric specification or size
- all studied nanotubes more susceptible to electrochemical dissolution than Pt(111) – potential corrosion problem

Acknowledgements

Thank you for you attention



LANL LDRD program for postdoctoral fellowship

U.S. Department of Energy, Energy Efficiency and Renewable Energy for financial support

National Energy Research Scientific Computer Center, Pacific Northwest
National Laboratory Advanced Computing Center

Neil Henson & Fernando Garzon
C. Taylor, J. Rossmeisl, P. Lazic