



# Atomistic Models for Scintillator Discovery

SAND2010-4906C



**SPIE**  
**Optics + Photonics**

**August 1-5, 2010**  
**San Diego, California, USA**

**Xiaowang Zhou, F. Patrick Doty, and Pin Yang**  
**Sandia National Laboratories**



# Two Problems of Current Interests

---

## ELPASOLITES:

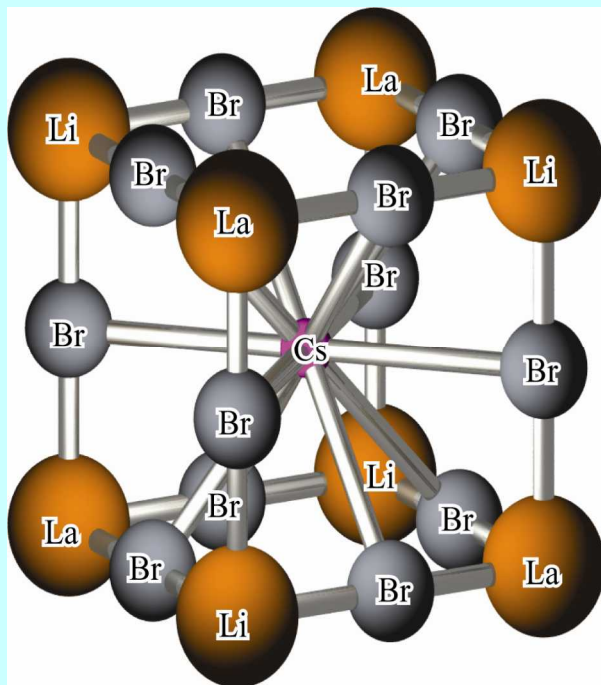
$A_2B\text{Ln}X_6$ : A, B: alkali; Ln: lanthanide; X: halogen.  
Cubic crystals have good properties and low cost.  
But how to design materials that have cubic crystals?

## LANTHANIDE (e.g., $\text{LaBr}_3$ ):

Extremely brittle, high cost, fracture in the field.  
What is the deformation mechanisms and can it be strengthened?

# Electronegativity Effect

## Double Perovskite $\text{Cs}_2\text{LiLaBr}_6$



- [1]. P. Yang, M. A. Rodriguez, F. P. Doty, X. Zhou, M. R. Sanchez, and K. S. Shah, submitted.  
[2]. X. W. Zhou, and F. P. Doty, *Phys. Rev. B*, **78**, 224307 (2008).

## Observations:

1. Lattice constant of  $\text{Cs}_2\text{LiLaBr}_6$  is  $a = 2r_{\text{LiLa}} = 11.289 \text{ \AA}^{[1]}$ , i.e.,  $r_{\text{LaLa}} = 7.983 \text{ \AA}$ .
2. fcc La has a lattice constant  $a = 5.307 \text{ \AA}$  and a cohesive energy  $E_c = -4.446 \text{ eV/atom}^{[2]}$ . In fcc La,  $r_{\text{LaLa}} = 3.753 \text{ \AA}$ .
3. It is difficult to for a potential to be transferrable to both  $\text{Cs}_2\text{LiLaBr}_6$  and La.

## Solution:

The electronegativity difference-induced ionization can increase the bond length and reduce the bond strength.

# Embedded Ion Method (EIM)

$$E = \frac{1}{2} \sum_{i=1}^N \sum_{j=i_1}^{i_N} \phi_{ij}(r_{ij}) + \sum_{i=1}^N E_i(q_i, \sigma_i)$$

system total energy

$$E_i(q_i, \sigma_i) = \frac{1}{2} q_i \cdot \sigma_i^*$$

embedding energy at i

$$q_i = \sum_{j=i_1}^{i_N} \eta_{ji}(r_{ij})$$

charge on atom i

$$\sigma_i = \sum_{j=i_1}^{i_N} q_j \cdot \varphi_{ij}(r_{ij})$$

electrical potential (in voltage) at i

$$E = \frac{1}{2} \sum_{i=1}^N \sum_{j=i_1}^{i_N} \phi_{ij}(r_{ij}) + \sum_{i=1}^N \left\{ \left( \sum_{j=i_1}^{i_N} \eta_{ji}(r_{ij}) \right) \cdot \sum_{j=i_1}^{i_N} \left[ \left( \sum_{k=j_1}^{j_N} \eta_{kj}(r_{jk}) \right) \cdot \varphi(r_{ij}) \right] \right\}$$

\*when  $\varphi_{ij}(r) \sim r^{-1}$ , the embedding energy reduces to Coulomb interactions between i and its neighbors:

$$E_i(q_i, \sigma_i) = \frac{1}{2} \sum_{j=i_1}^{i_N} \frac{q_i \cdot q_j}{r_{ij}}$$

# The 1<sup>st</sup> Problem: Crystal Prediction

$A_2BLnX_6$ : A, B: alkali; Ln: lanthanide; X: halogen.

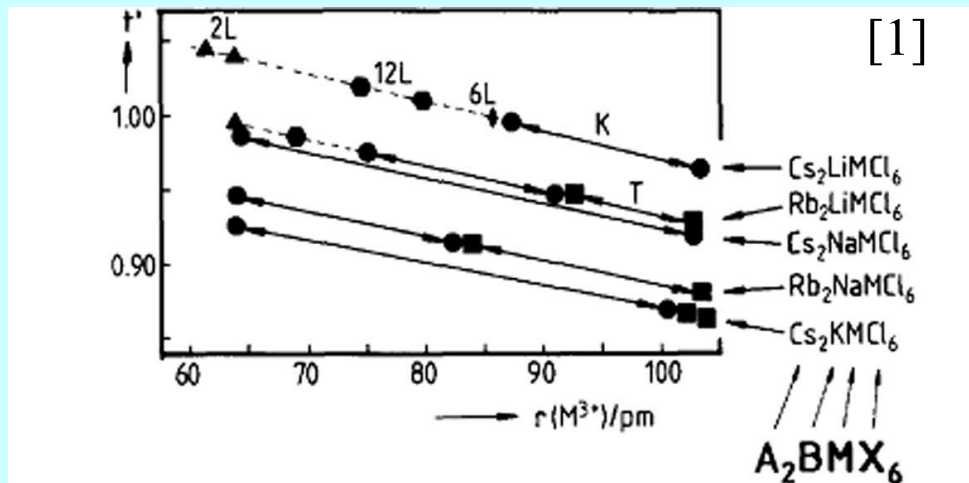
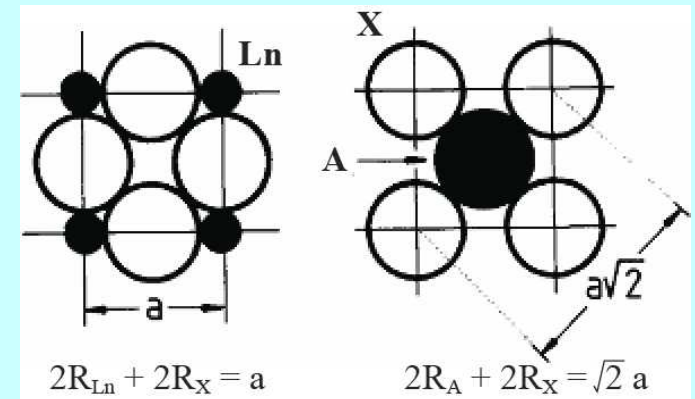


Fig. 11. The derivation of Goldschmidt's tolerance factor  $t'$  (above) and its (useless) application to chloro-epasolites [1]. G. Meyer, *Prog. Solid St. Chem.*, **14**, 141 (1982).



$$t' = \frac{R_A + R_X}{\sqrt{2} \left( \frac{R_B + R_{Ln}}{2} + R_X \right)}$$



# Two Notions

**1. Goldschmidt criterion<sup>[1,2]</sup> (hard sphere model) constrains bond length  $r$  to ionic radii  $R$ :  $r_{AA} = 2R_A$ ,  $r_{BB} = 2R_B$ ,  $r_{AB} = R_A + R_B$ , ... Our model treats all bond lengths as independent parameters.**

**2. We model alkali halides as a first step because elpasolites are composed of alkali, halogen, and lanthanide elements.**

[1]. V. M. Goldschmidt, *Geochemische Verteilungsgesetze der Elemente*. Norske Videnskap, Oslo, 1927;

[2]. L. Liang, L. Wencong, and C. Nianyi, *J. Phys. Chem. Sol.*, **65**, 855 (2004).

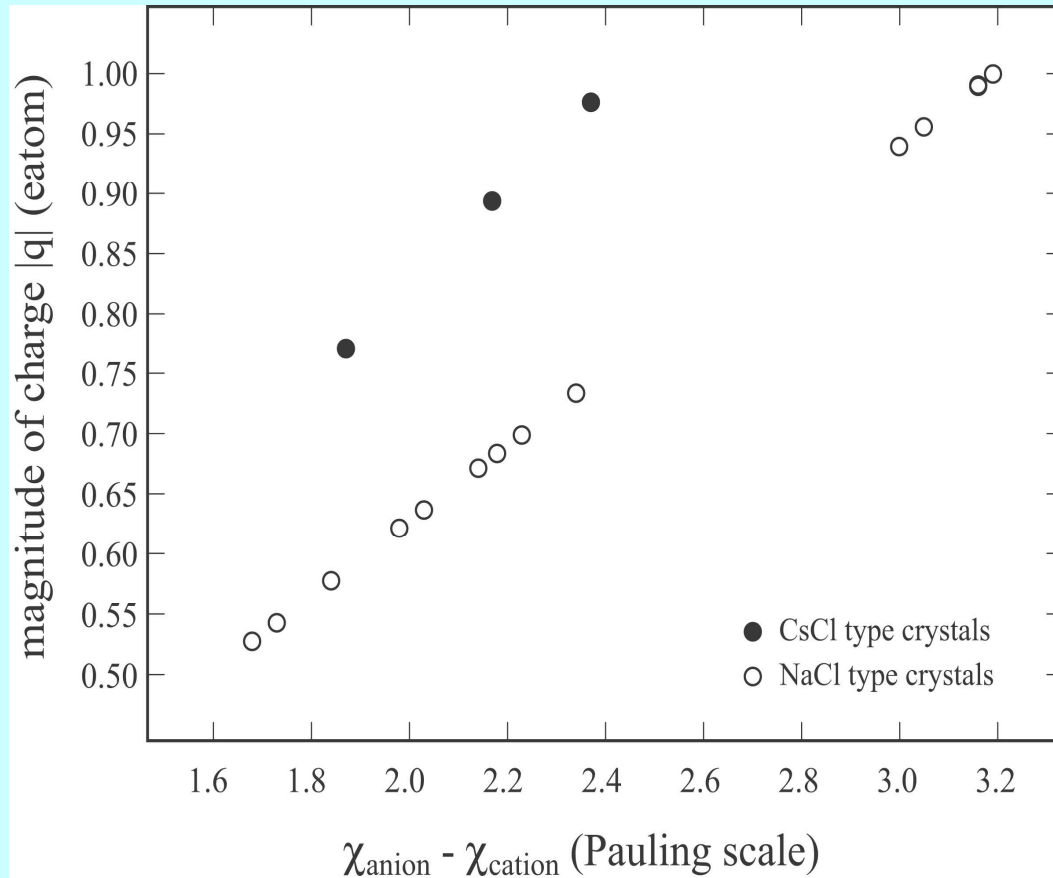


# EIM Potential Database

---

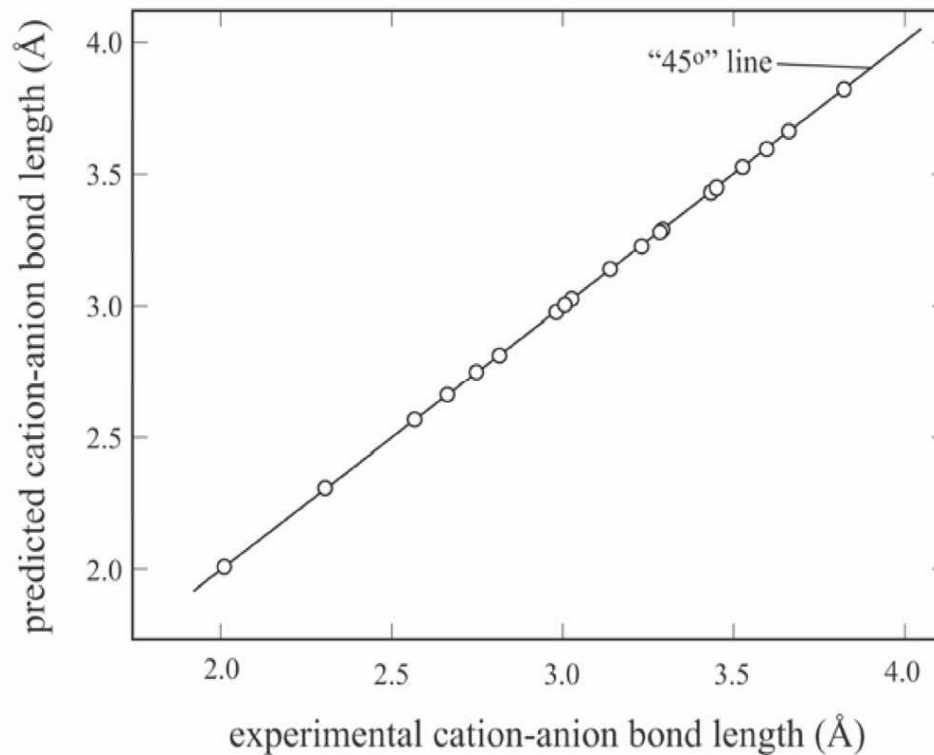
- 1.contains all nine alkali halide elements (Li, Na, K, Rb, Cs F, Cl, Br, I),**
- 2.focuses on atomic size, but also considers electronegativity and bond energy effects,**
- 3.uses elemental properties directly as model parameters without parameterization, and**
- 4.gives good trend of charge, energy, and bond length predictions.**

# Charge vs. Electronegativity

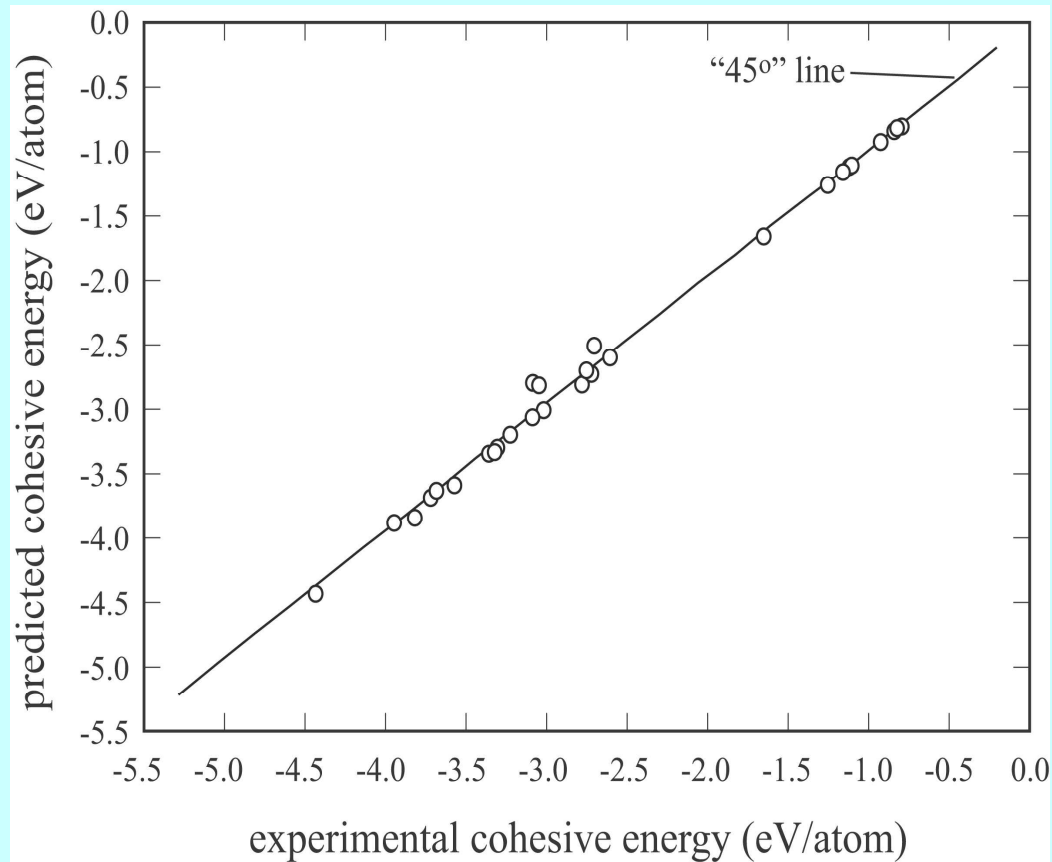




# Lattice Constant Prediction

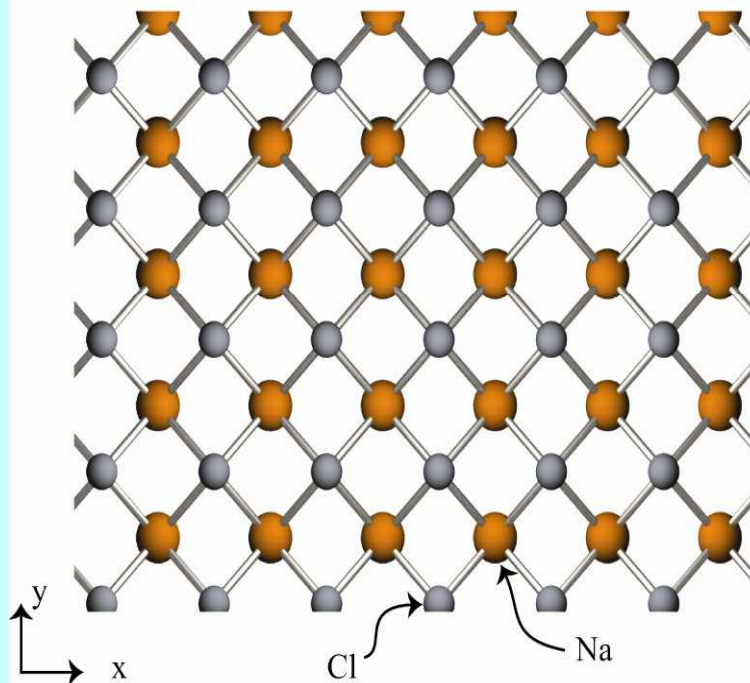


# Cohesive Energy Prediction

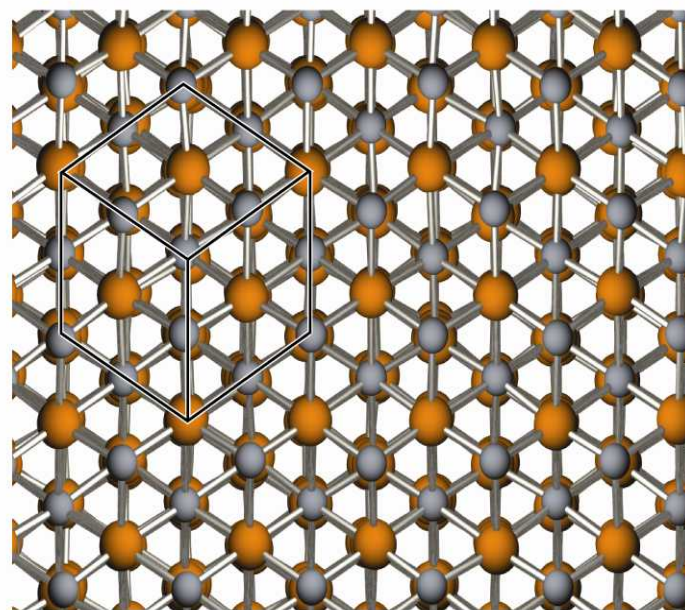


# Structure Prediction

*(a)* initial “CsCl” crystal structure of NaCl

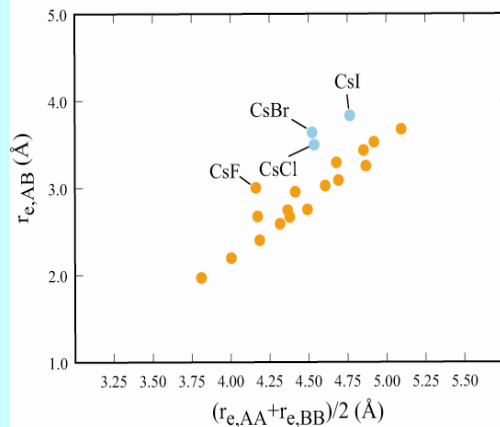


*(b)* after 10 ns 1400-to-300 K annealing

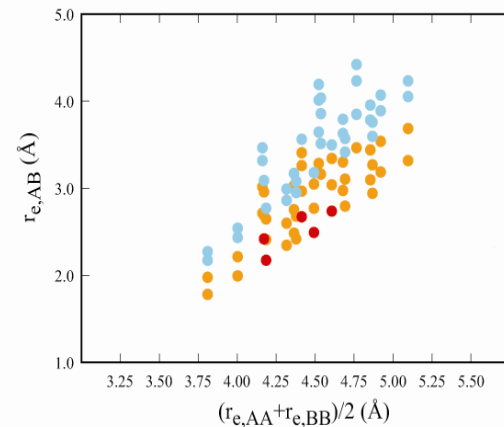


# Crystal Phase Diagram

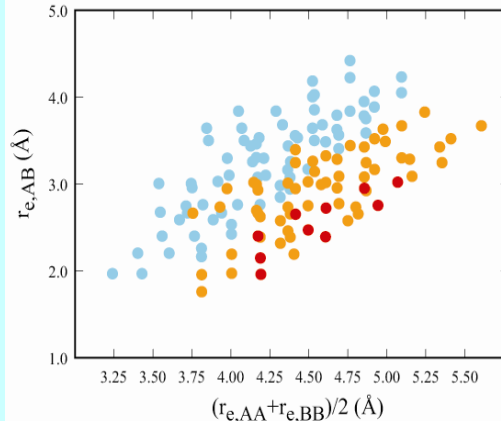
(a) 20 alkali halide structures



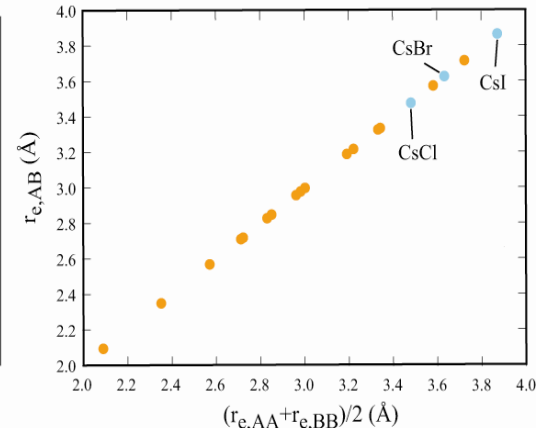
(b) varying  $r_{e,AB}$



(c) varying  $(r_{e,AA} + r_{e,BB})/2$



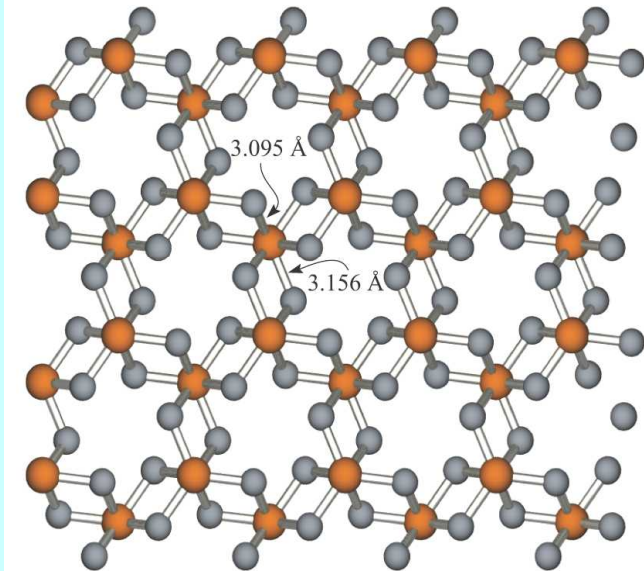
(d) the hard sphere model



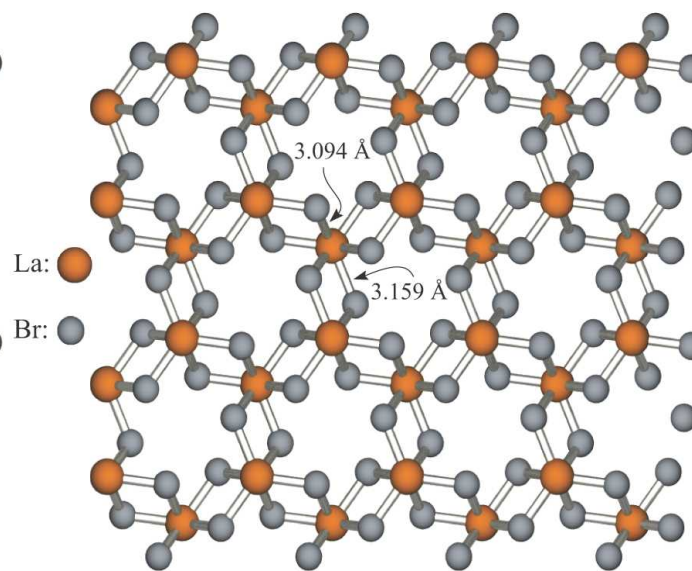
# The 2<sup>nd</sup> Problem: LaBr<sub>3</sub> Fracture

(a)  $c/a \sim 0.57$ ; (b) rotated Br polyhedron; (c) two La-Br bond lengths; and (d) big lattice hollows

(a) experimental LaBr<sub>3</sub> structure



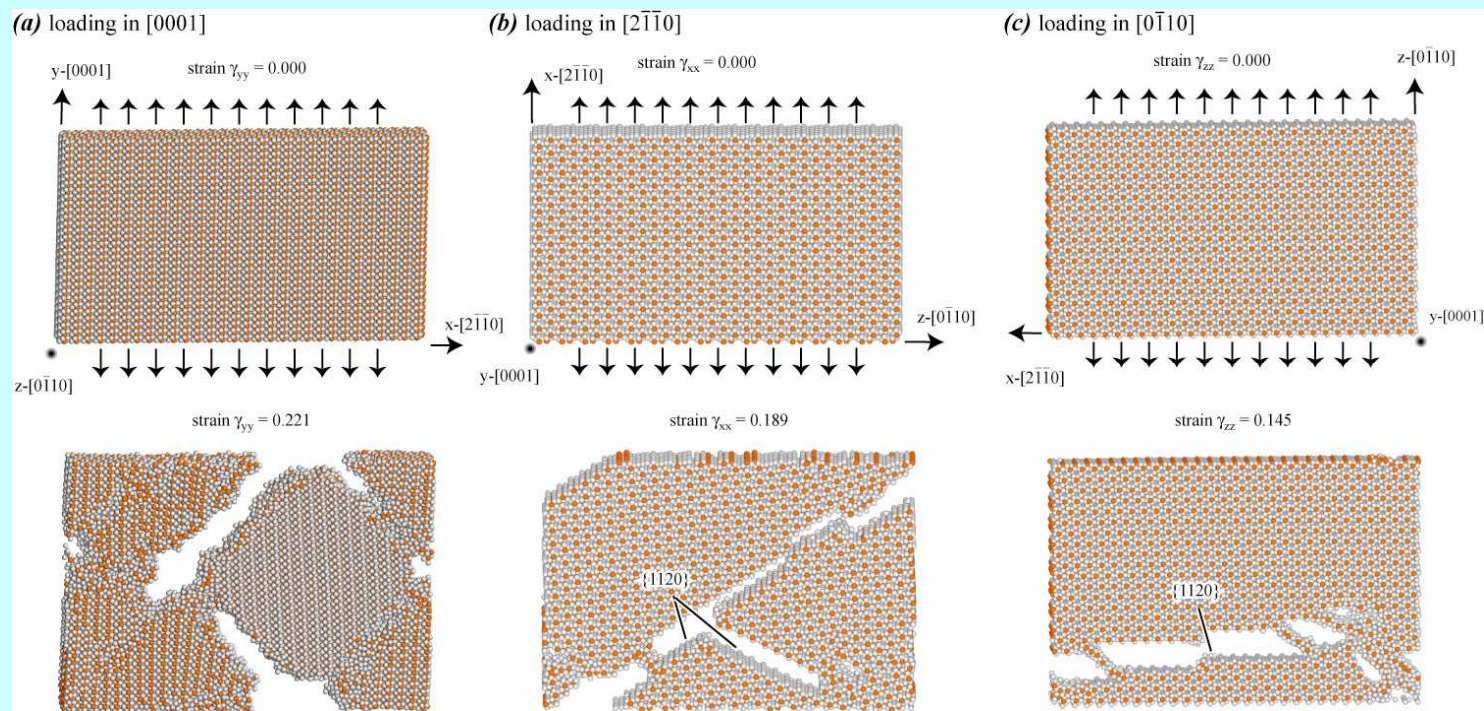
(b) predicted LaBr<sub>3</sub> structure



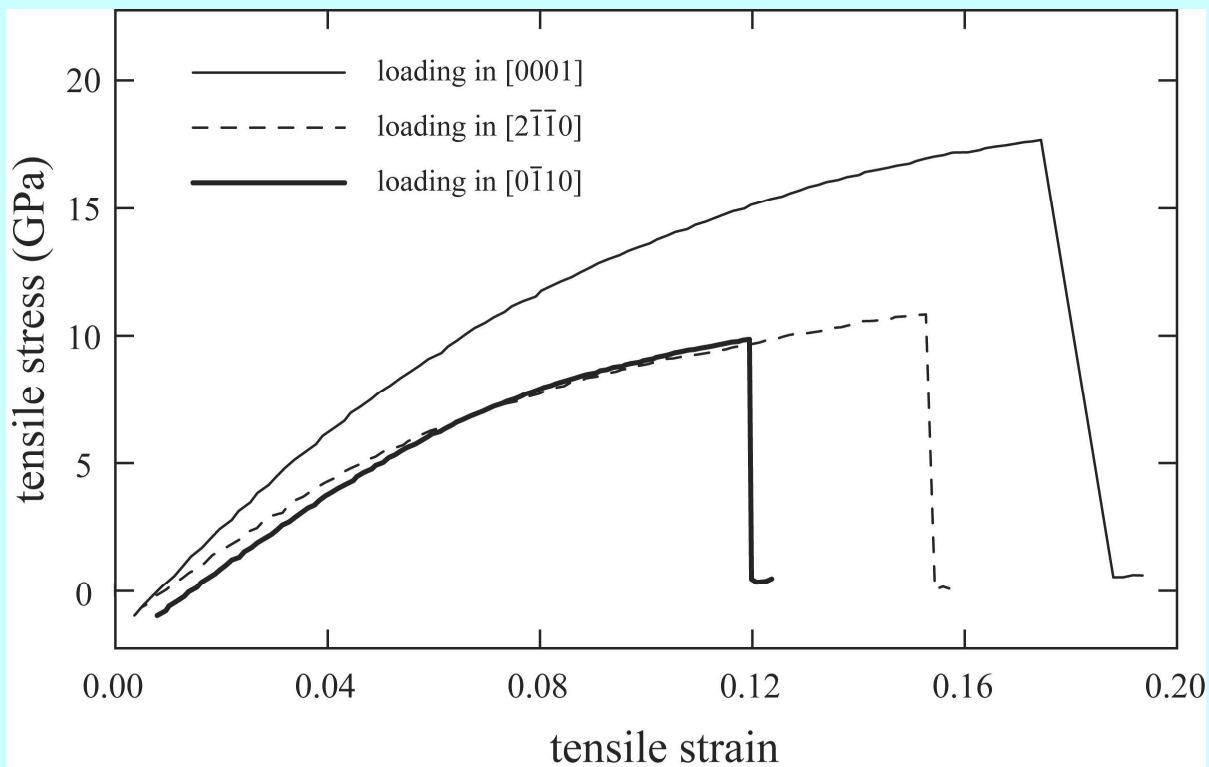
Our LaBr<sub>3</sub> EIM captures the structural properties!



# Tensile Fracture vs. Orientation

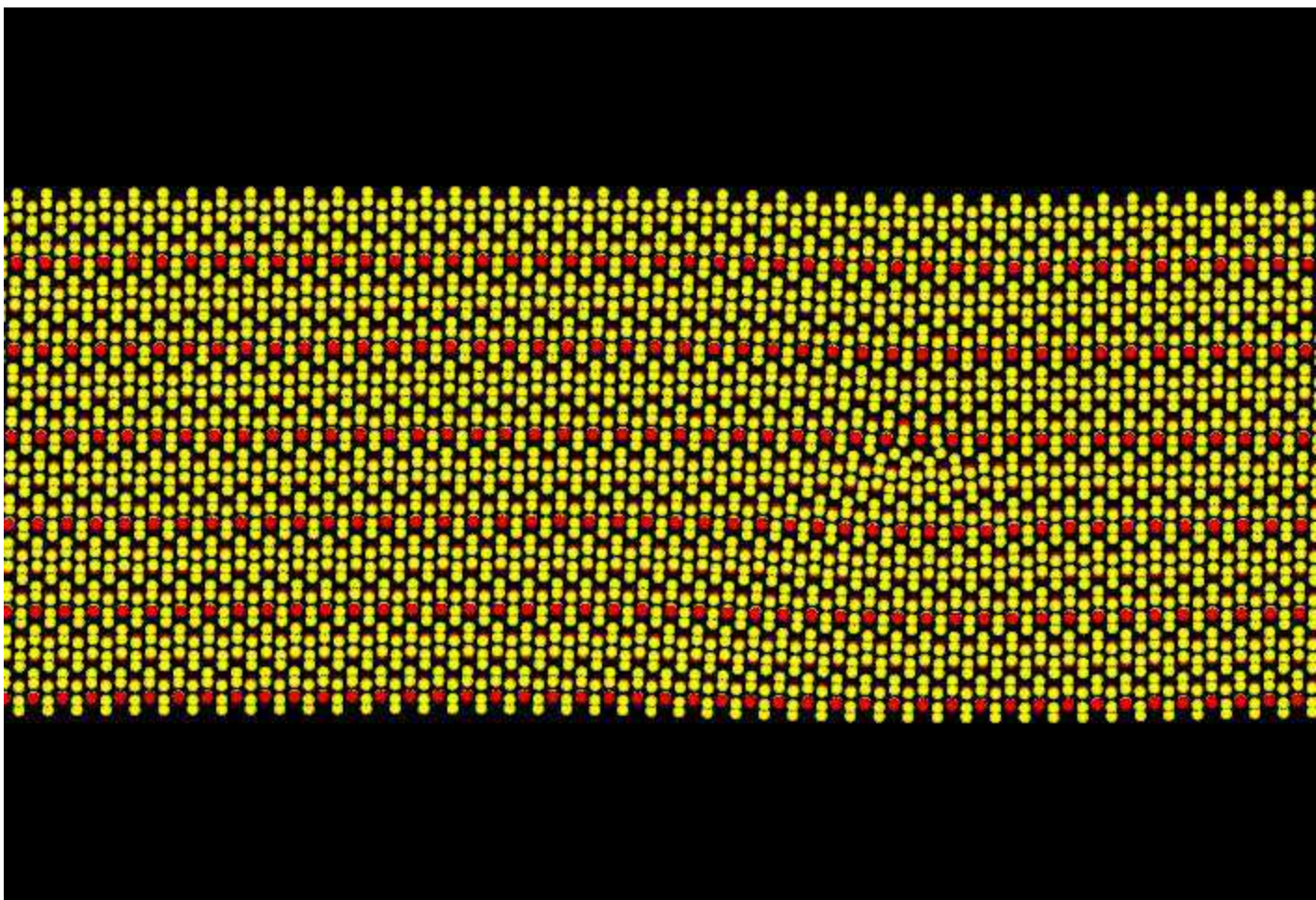


# Tensile Stress vs. Strain



# [0001] Edge Dislocation under 4GPa at 0 K

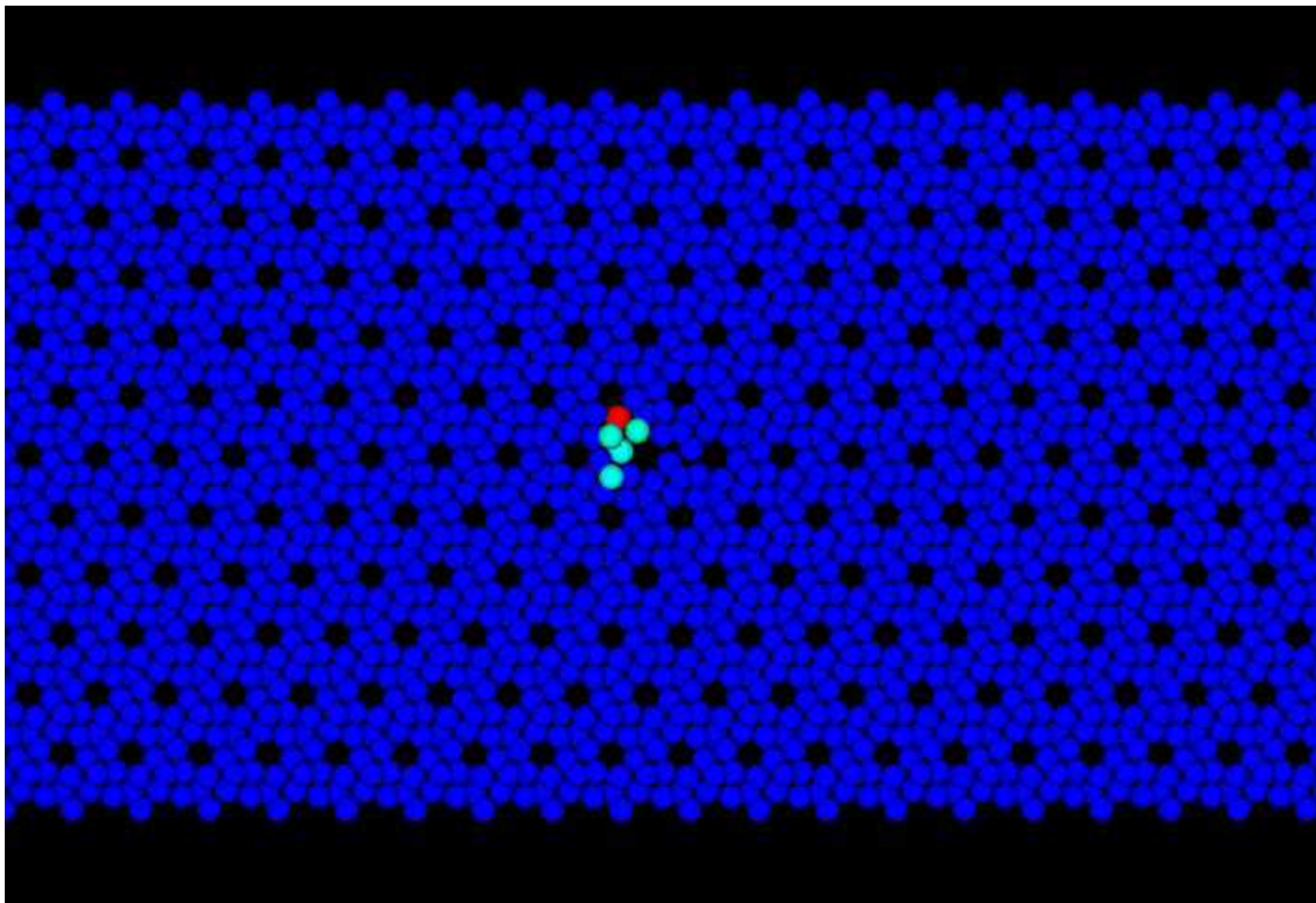
---





## [0001] screw dislocation under 3GPa at 0 K

---





# Slip Systems

---

- The only mobile dislocations in  $\text{LaBr}_3$  are on the prism planes.
  - The material cannot be strain-hardened.
- Approaches that block dislocation motion on the prism planes must be sought.



# Conclusions

---

- **New atomistic simulation methods enable crystal rules to be explored in atomic size-electronegativity-bond energy space.**
- **A new crystal rule beyond the hard sphere model was discovered for alkali halides.**
- **The critical stress for the  $\{1120\}$  cleavage is very low in  $\text{LaBr}_3$ .**
- **The only mobile dislocations in  $\text{LaBr}_3$  are on the  $\{1120\}$  prism planes.**
- **$\text{LaBr}_3$  cannot be strain-hardened. Approaches that block the dislocations (e.g., solid-solution strengthening) can be used.**



# Acknowledgement

---

**Sandia National Laboratories is a multi-program laboratory operated by Sandia Corporation, a wholly owned subsidiary of Lockheed Martin company, for the U.S. Department of Energy's National Nuclear Security Administration under contract DE-AC04-94AL85000. This work is supported by the NNSA/DOE Office of Nonproliferation Research and Development, Proliferation Detection Program, Advanced Materials Portfolio.**