

Trends in Microjoining 2010

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Microjoining Landscape

- **Microelectronics**
 - Computers
 - Cell phones
 - Pacemakers
- **MEMS (adds mechanical functions)**
 - Sensors
 - Actuators
- **MOEMS (adds optical functions)**
 - Optical sensors
 - Optical switching
 - HDTV
- **Microfluidics (adds liquid flow functions)**
 - Chemlab on a chip
 - Medical implants
- **Photonics (all optical)**



Definition: Microjoining

- Two words, one word or hyphenated???
- JIS Z 3001 takes an operational approach and describes micro-welding as a
 - "generic term for welding processes which are applied to the sections which, owing to the object for welding being extremely small and/or fine, are affected by factors such as weldability, diffusion thickness, deformation amount and surface tension, particularly whose effect on dimensions is to be taken into account."
- K. Matsuyama chose "to treat the welding processes for components of about several millimetres or smaller as 'micro-welding'."



Definition: Microjoining II

- Several possible physical approaches:
 - Dimensions of objects being joined
 - One dimension, two, or all three?
 - One object or all objects (i.e. thin to thick)?
 - Filler material added (thickness/dia. or mass μg)
 - Thermal diffusion distance
 - Energy usage
 - $\mu\text{J}/\text{pulse}$ for spot welds
 - $\mu\text{J}/\mu\text{m}$ for seam welds
 - Timescale
 - μs
 - Ratio of surface tension/other significant forces ~ 1
- For our purposes, meeting *any* of the above will suffice.



Microjoining Processes

- Thermocompression/thermosonic/ultrasonic
 - Ball/wedge bonding
 - Flip chip (Au stud bump)
 - also TAB, beam lead
- Flip Chip reflow of solder ball arrays
- μ adhesive bonding
- μ e⁻-beam
- μ laser
- Focused ion beam (FIB) deposition
- μ spark (μ arc welding)
- μ resistance welding
- Exothermic nanolaminates



Microjoining Process Attributes

- Visualization of parts/process difficult (tiny dimensions)
- Handling/fixturing difficult (small, fragile, floppy)
- Control of process difficult (small energy inputs/short timescales, precise path alignment)
- Reproducibility critically dependent upon reproducible parts/fitup
- Sensitive to surface effects
 - Contamination (extreme surface/volume ratio)
 - Surface tension forces
- Difficulty inversely proportional to size!
- Poor productivity except:
 - Laser with scanner-type beam positioning
 - Ebeam with rapid magnetic lens beam control\
 - Parallel processes



Microjoining & Materials

Similar and dissimilar joints between:

- **Metals (join by all processes)**
 - Au, Al, Cu, Si, Ni, Kovar, Stainless Steel, Elgiloy, Nitinol
- **Transparent materials (not RW-US-Arc, eB difficult, LB with special techniques, FIB, At. diff.)**
 - Sapphire, glass, SiO₂, other photonic materials
- **Polymers (not Solder-eB-Arc)**
- **Ceramics (metallization & solder/braze, adh, FIB)**
- **Nanomaterials (carbon nanotubes, nanowires: eB, LB, FIB, solder, adh, RW + W catalyst, At. diff.)**
- **Meta-materials (negative refractive index nanocomposites, unknown durability: adh, FIB, low temp solder?)**



Macro vs. Micro Comparison of Various Origin Stress/Force Magnitudes Acting on a Weld Pool

Force origin:	Macro magnitude: pressure/stress	Macro magnitude: force	Micro magnitude: pressure/stress	Micro magnitude: force
Thermal expansion (M) (ambient-to-molten)	6.5 GPa	6.5 kN	6.5 GPa	0.65 N
RW electrode inertial melting (P) solidifying (P)		0.02-2 N 0.13-13.3 N		0.05-0.5 N 0.1-10 N
Evaporation (P)	1.5-7 kPa	1.5-7 mN	1.5-7 kPa	0.15-0.7 μ N
Surface tension (M)	2 kPa	2 mN	200 kPa	20 μ N
Marangoni shear (M)	70 Pa	70 μ N	0.7 MPa	70 μ N
Impinging droplets (P)	2.7 kPa	12 mN	67 kPa	3 mN
Lorentz vortex (P)	3.7×10^{-4} N/mm ³	37 mN		N. A.



Macro vs. Micro Comparison of Various Origin Stress/Force Magnitudes Acting on a Weld Pool II

Pool sloshing (M)	200 mN		$20 \mu\text{N}$	
Stage motion, centrifugal (P)	44 mN		44 nN	
Liquid viscosity shear (M)	6 Pa	$6 \mu\text{N}$	6 Pa	0.6 pN
Aerodynamic shear stress (P)	70-400 Pa	$70-400 \mu\text{N}$	70-400 Pa	$7-40 \text{ pN}$
Aero stagnation pressure (P)	0.4-600 kPa	$0.4-600 \text{ mN}$	0.4-600 kPa	$0.04-60 \mu\text{N}$
Lorentz GMAW pinch (P)	6 kPa	6 mN	N.A.	
Current carrying conductor "kick" (P)	200 N/m <i>(1 m cable)</i>	200N	80 mN/m <i>(10 cm cable)</i>	8 mN
Metallostatic head (M)	0.8 kPa	0.8 mN	20 Pa	$0.002 \mu\text{N}$
Buoyancy force (M Δpg): Autogenous ⁺ Dissimilar [#]	2.1 kN/m ³ 52 kN/m ³	21 mN 52 mN	2.1 kN/m ³ 52 kN/m ³	$2.1 \mu\text{N}$ $5.2 \mu\text{N}$



Ball/Wedge Bonding

- **Mature technology, around since 1960's**
- **Incremental progress:**
 - **Faster bond cycle (few 10's of ms, under CNC)**
 - **Better placement (finer pitch)**
 - **Better loop height control (multilevel wire tiers)**
 - **Improved process yield (~25 ppm defect rate)**
 - **Au, Al, Cu (and other) wires**
 - **Broader wire diameter range, insulated wires**
 - **Stud bumping**
 - **3D (vertically-stacked dies)**



Ball Bonding

- **from** <http://www.youtube.com/watch?v=pajE4Bi6Xts&feature=related>
- **search "Gold Ball Bonding"**

QuickTime™ and a decompressor are needed to see this picture.



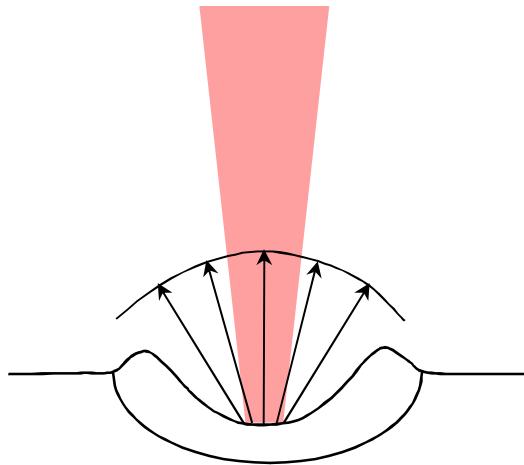


μ Laser Welds

- **Hardware is becoming readily available**
 - Near diffraction limit focused beams available from single mode fiber lasers
 - Sub-micron-resolution CNC stages
 - Rapid motion beam scanners
 - Digital optics beam splitters
- **Questions:**
 - Welding vs drilling at small spot sizes
 - Beam characterization at small spot sizes
 - Fixturing / handling!!!



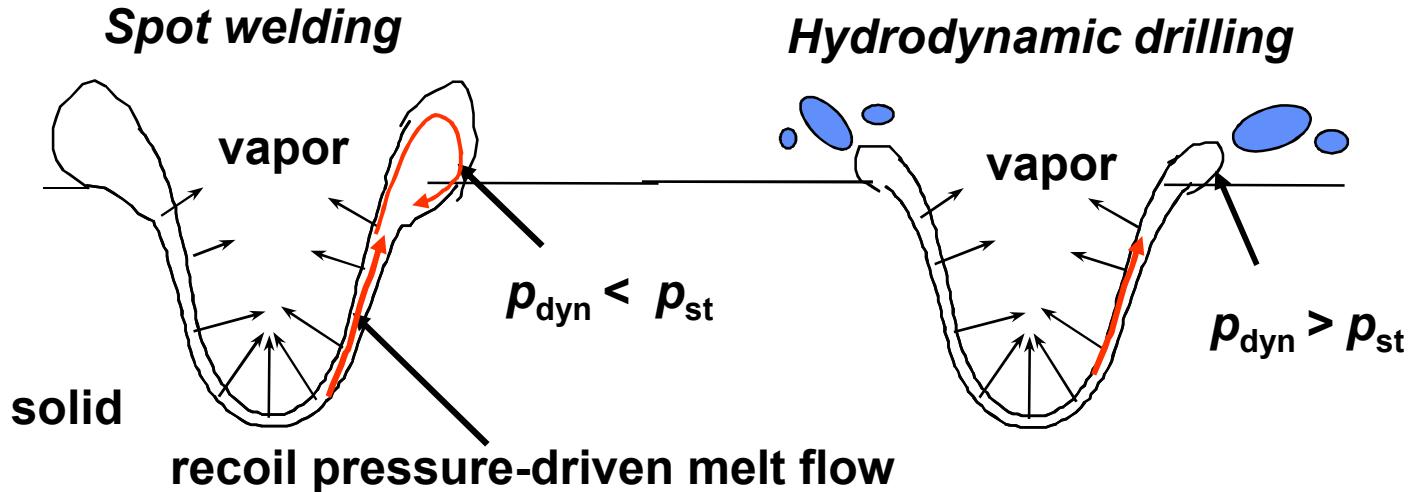
Forces Acting in Laser Welding



Evaporation:	7000 Pa
Surface tension	
Driving:	700
Resisting:	2000
Viscosity:	60
Momentum/Inertia:	0.010

Approximate values for forces acting in mm-size welds; only surface tension resisting force from capillarity changes ($\sim 1/r$) with size

Welding vs Drilling



$$p_r = 0.54 B_0 (T_s)^{-1/2} \exp(-U/kT_s)$$

Values:
1.5 (meas.) - 7 kPa (3D calc.)

B_0 = evaporation constant

T_s = surface temperature

U = latent heat of evaporation/atom

k = Boltzman's constant



Welding vs Drilling Initial Model

$$t_{ej} = \frac{r_m}{\langle v_m \rangle} \leq \tau - t_m$$

Melt displacement criterion, where:

t_{ej} **time for melt displacement**

r_m **melt pool radius**

$\langle v_m \rangle$ **melt velocity average**

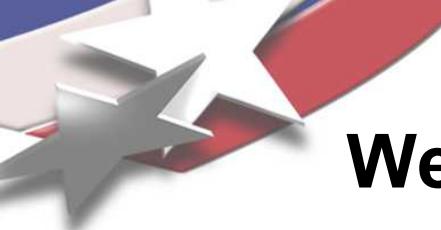
τ **laser pulse duration**

t_m **time to the initiation of surface melting**

+ conservation of momentum, energy + BC's →

calculated displacement thresholds implied:

Small diameter pulses should drill immediately upon surface melting, particularly for longer τ !



Weld vs Drill Experimental Results

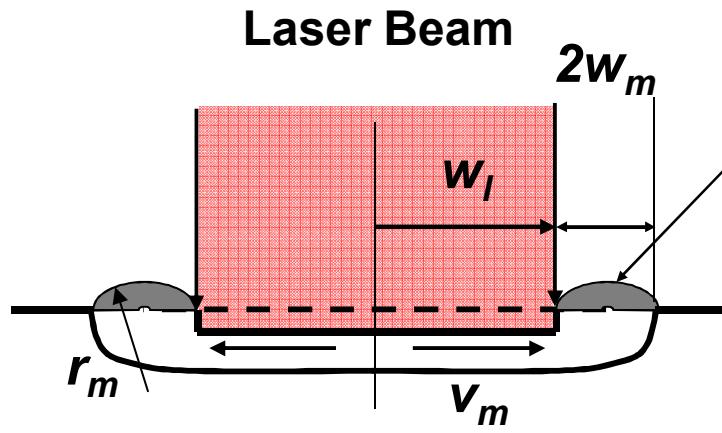


Ni 200
pulsed Nd:YAG laser
5 ms pulse
20 μm radius beam

Results showed it is possible to make small welds without drilling and that a progression from flat to concave to drilled occurs with increasing energy



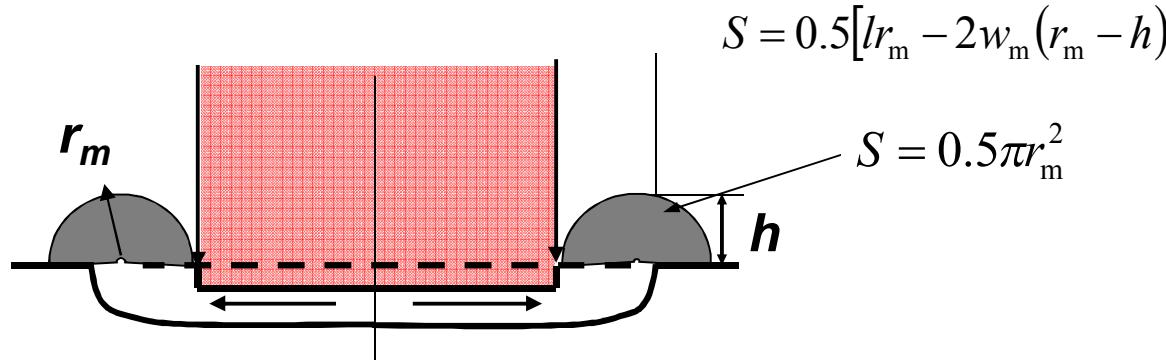
Improved Model: Including Surface Tension



$$S = 2w_m \int_0^t v_m dt$$

$$h = r_m - \sqrt{r_m^2 - w_m^2}$$

$$l \approx \sqrt{2w_m^2 + \frac{16}{3}h^2}$$

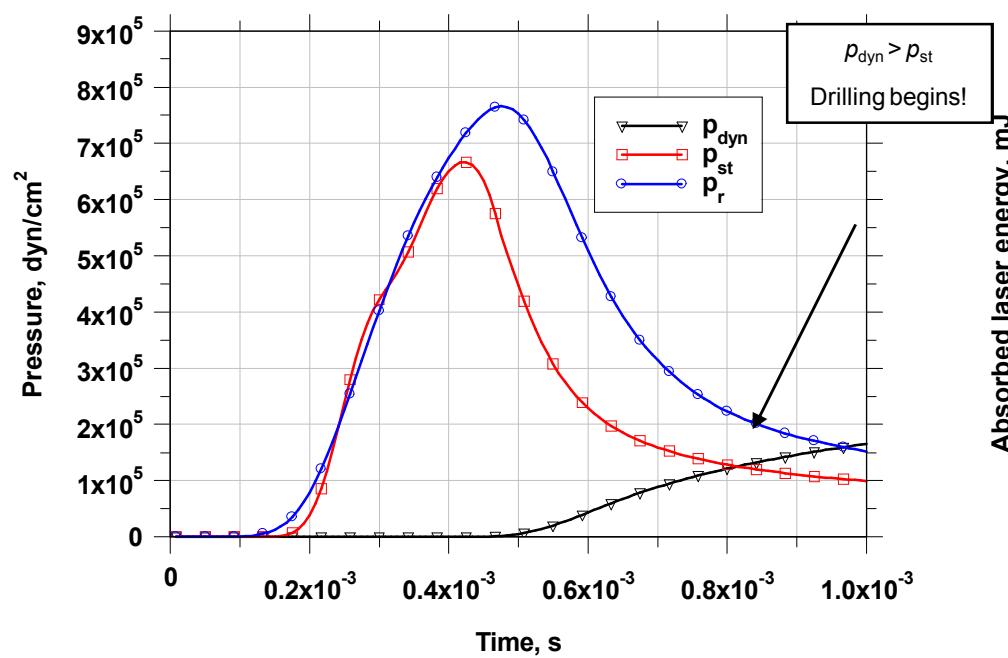


$$S = 0.5[lr_m - 2w_m(r_m - h)]$$

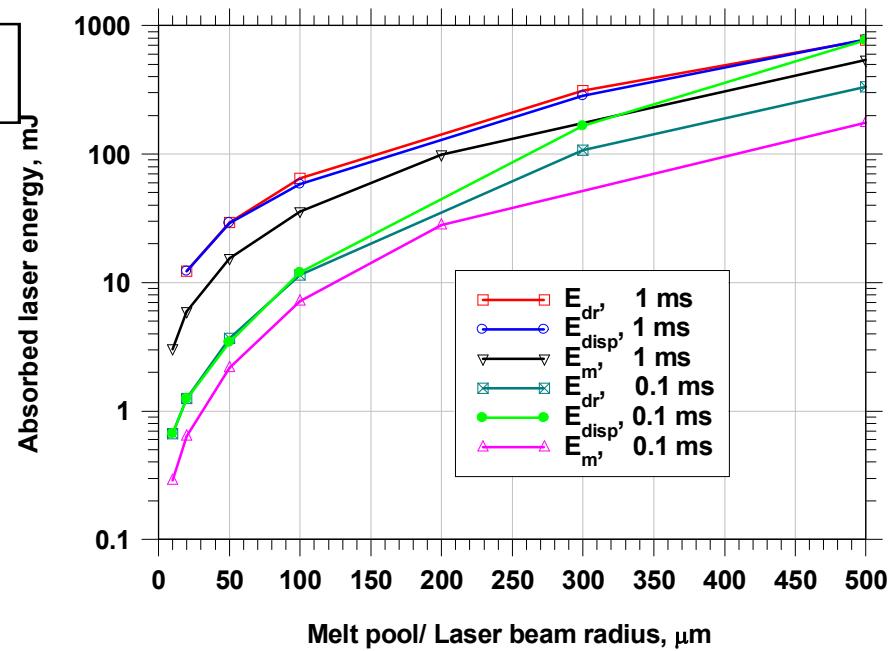
$$S = 0.5\pi r_m^2$$

Results

330 mJ, 1ms, 300 μ m radius
drilling pulse

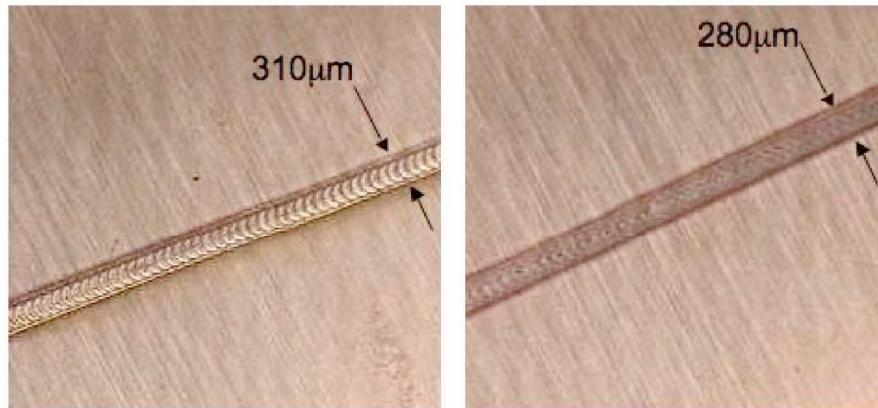


Surface tension retards the
displacement/drilling
threshold to > melting





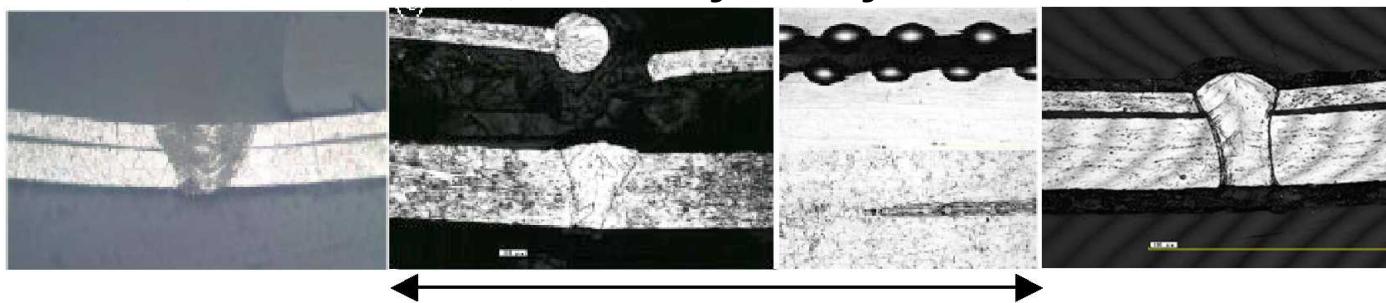
Examples of μ LBW



Top surface

Bottom surface

Cu, SHG Nd:YAG, courtesy of Miyachi Technos



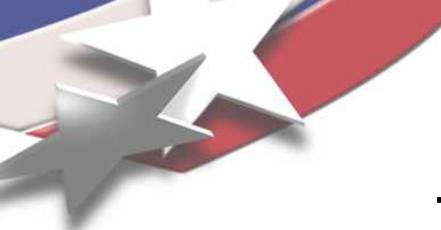
(a)

(b)

(c)

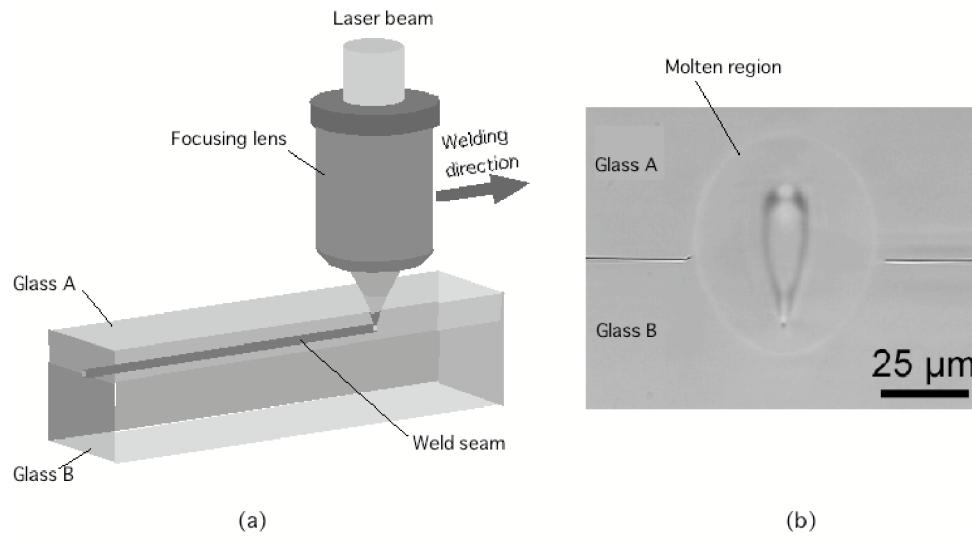
Single mode fiber laser: 10 μ m spot size, stainless steel foil lap welds
(a): 20 μ m/30 μ m, 25W @ 1m/s, (b): 10 μ m/30 μ m, 30 μ m gap not bridged,
(c): same as b, with no gap

Miyamoto, Park, Seo-heong, Ooie, Proc. 4th LPM



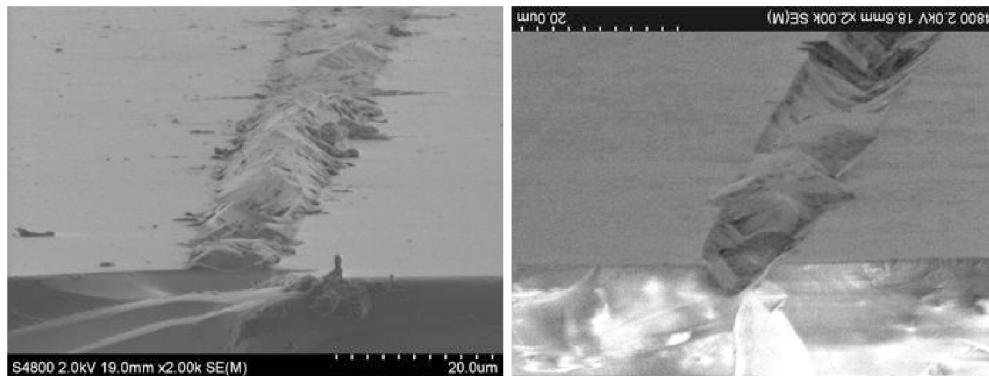
Example of μ LBW: Glass

from Miyamoto & Herrman Proc LPM2007

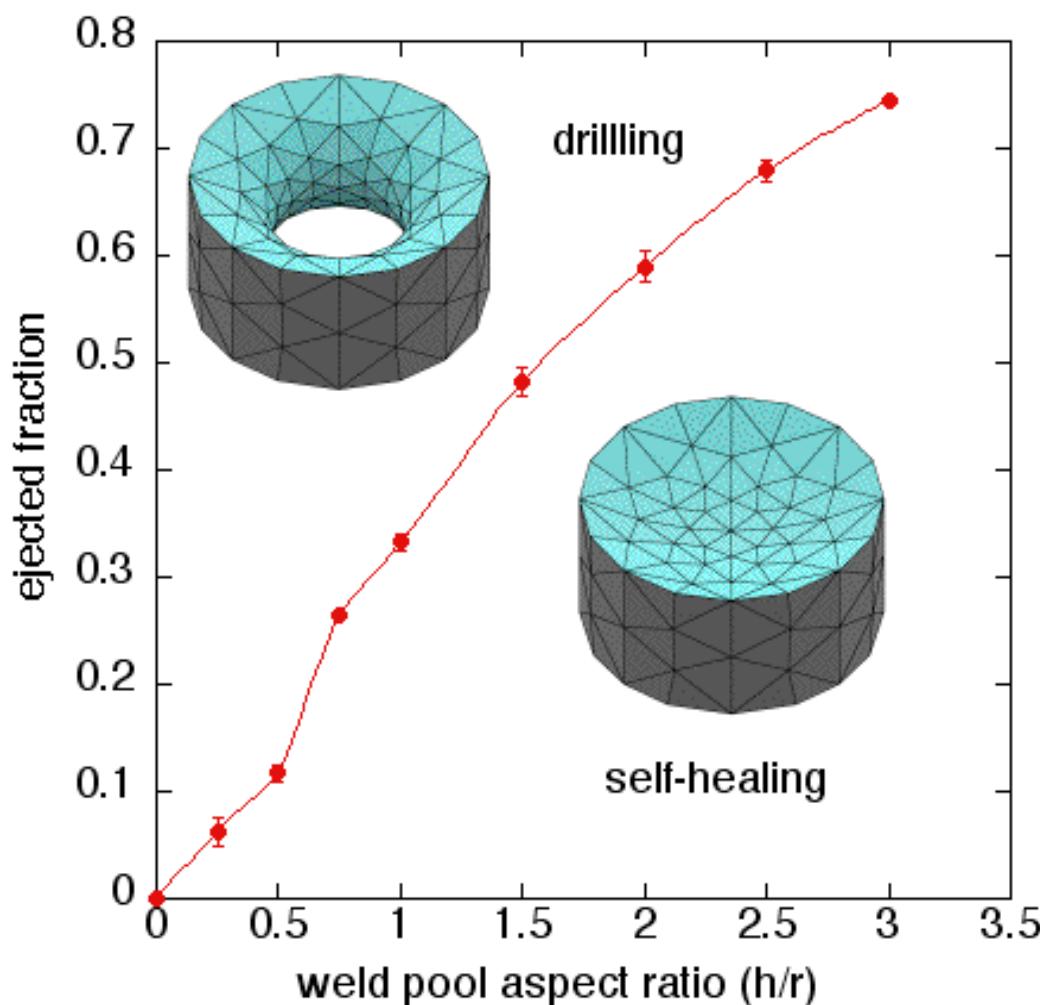


(a)

(b)



Surface Energy Considerations

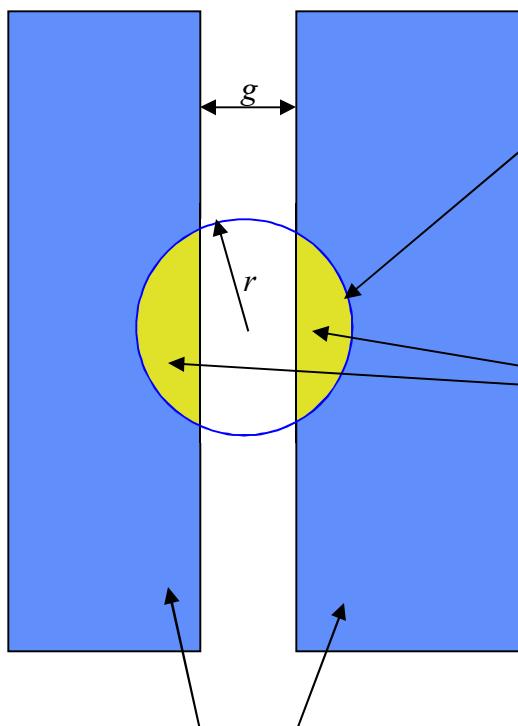


Scenario:
Perform lap spot weld in thin sheets where metal ejection occurs.

How much can be lost without leaving a hole?

Brakke, K.E., The Surface Evolver, available from
www.susqu.edu/facstaff/b/brakke/evolver/

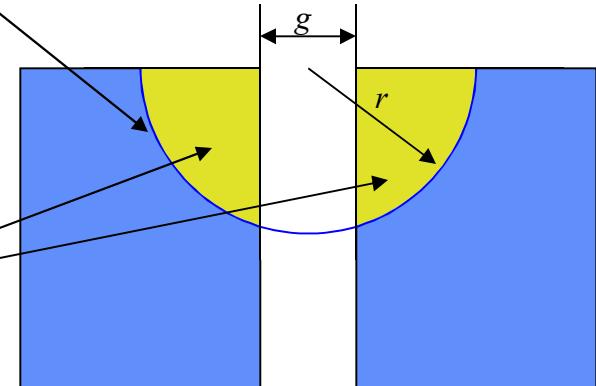
Surface Energy Consideration II



laser/e-beam hot zone
hemispherical $r = 1$
Material inside hot zone melts;
remainder stays solid

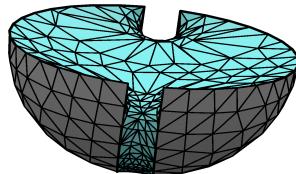
liquid (melt zone):
total volume, $v = (\pi/3)(2r+g)(r-g)^2$
contact angle $\theta = 10^\circ$

We begin with liquid contiguous
across the gap. Total liquid volume
remains constant as pool shape
evolves.

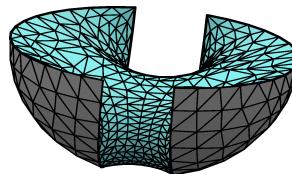


material thickness $> r$

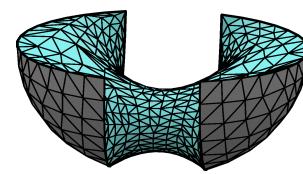
At a critical gap-to-width ratio the ligament separates:



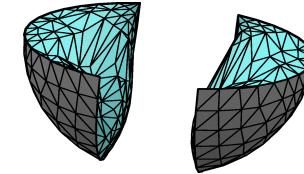
$g/r = 0.125$



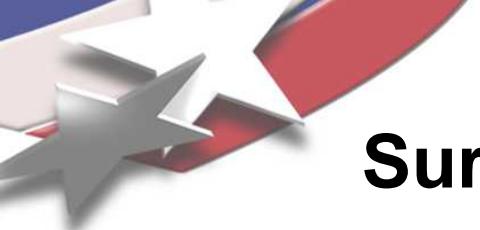
0.25



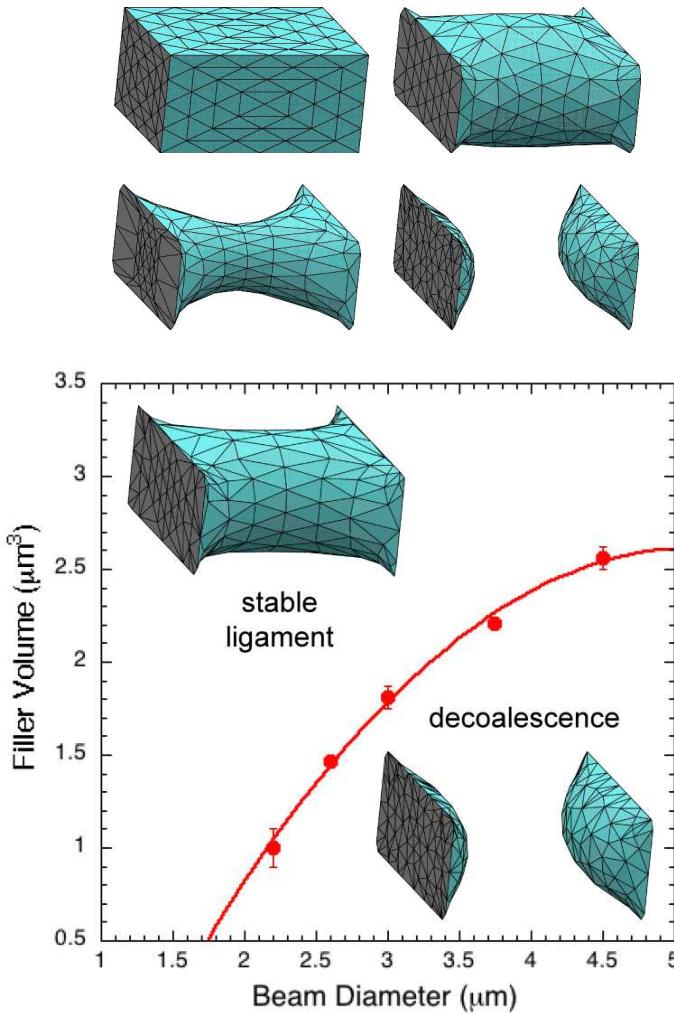
0.33



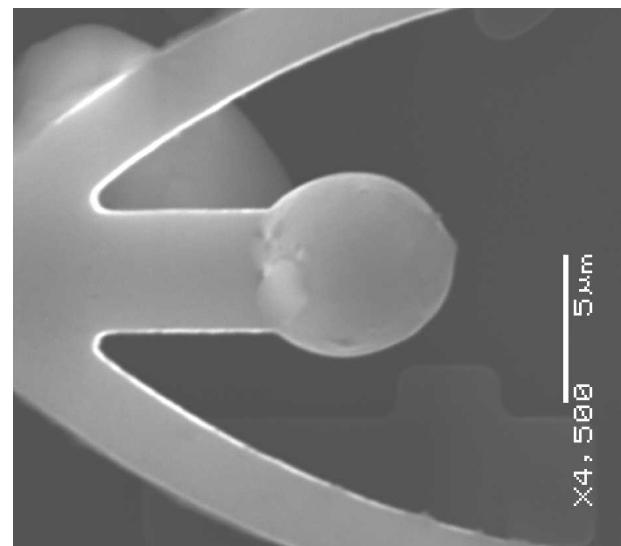
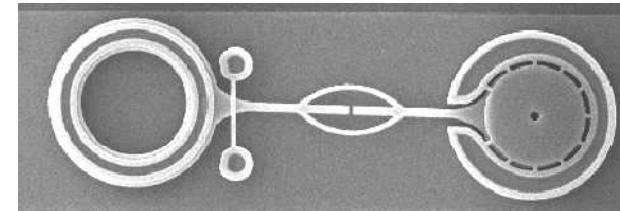
0.345



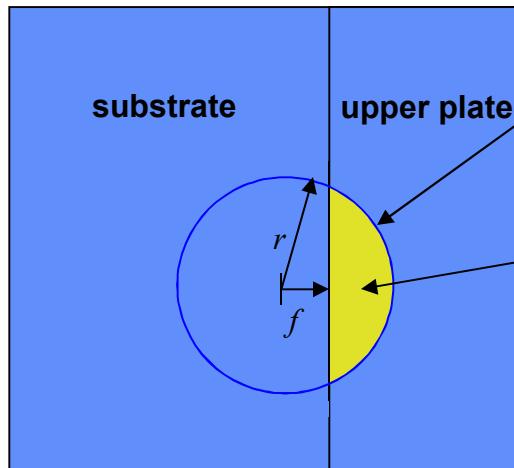
Surface Energy Considerations III



**Minimum melt volume
needed to bridge gap**



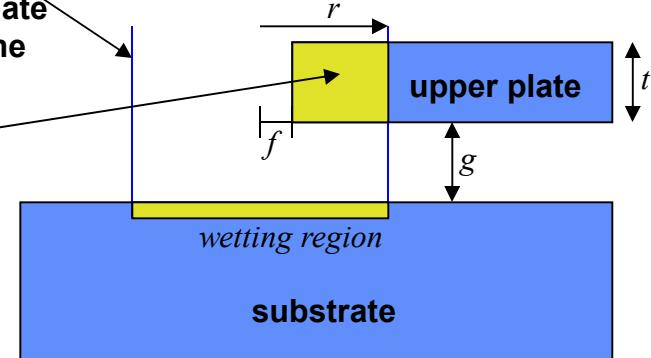
Surface Energy Considerations IV



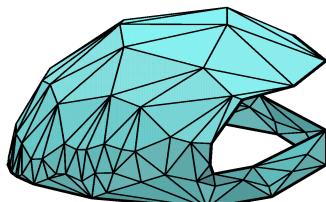
hot zone
 circular with radius $r = 1$
 may be offset from center
 Material inside the spot in the upper plate melts; that outside the spot and on the substrate remains solid.

liquid (melt zone):
 total volume, $v = t (r^2/2)(\varphi - \sin \varphi)$
 where $\varphi = 2\cos^{-1}(f/r)$

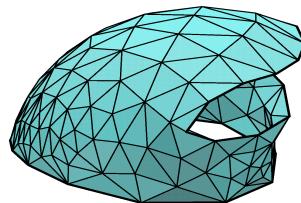
Contact angle varies. The total liquid volume remains constant as the liquid pool shape evolves. Liquid wets the substrate hot spot and the upper plate's hot edge only.



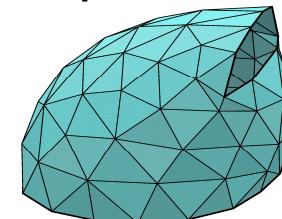
As the gap becomes larger, contact area with the upper plate decreases; eventually at $g \sim 0.53 \pm 0.03$, the droplet decoalesces



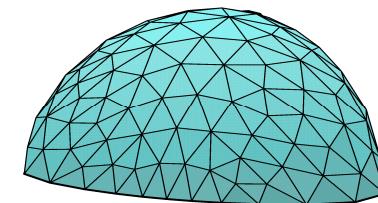
gap width = 0.125



0.25



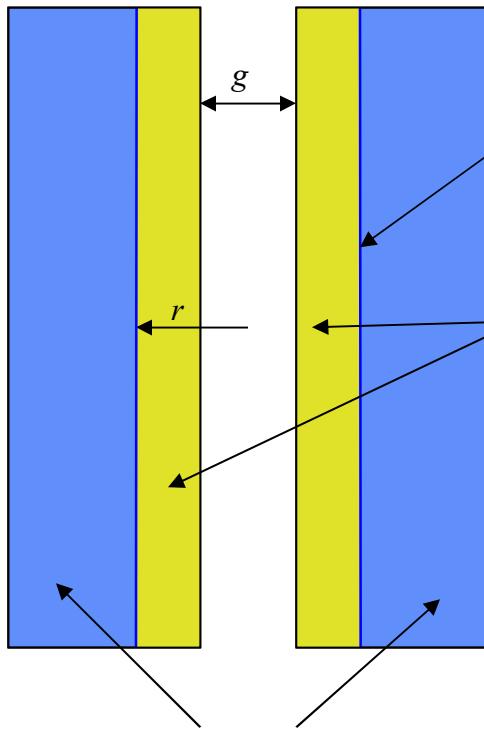
0.5



0.5625

Notes: For this system, spot radius = 1, upper plate thickness = 0.25, contact angle = 90° offset = 0. Only the liquid is shown. Total volume of liquid is constant.

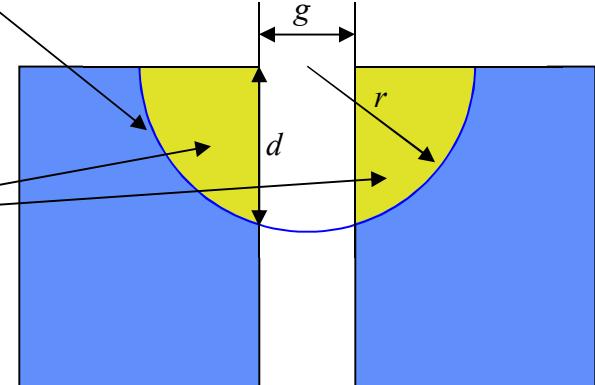
Surface Energy Considerations VI



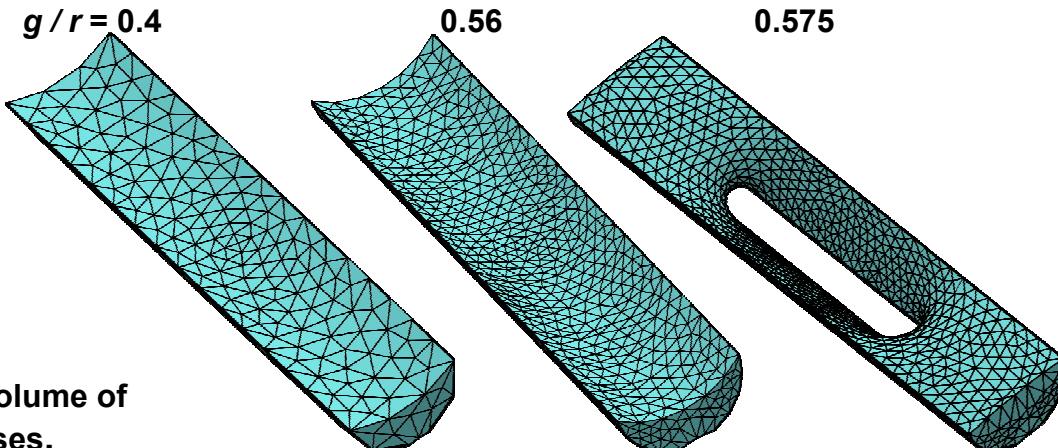
weld line (hot zone):
 hemicylindrical with radius $r = 1$
 centered between the bars
 Metal inside the hot zone melts; metal
 outside the hot zone remains solid.
 The maximum depth of the weld pool

$$d = (r^2 - g^2/4)^{1/2}$$

liquid (melt zone):
 total volume, $v = (\pi r^2 l/2) - (dgl/2) - [r^2 l \arcsin(g/2r)]$
 liquid/solid contact angle, $\theta = 10^\circ$
 We begin with liquid contiguous across the
 gap. The total volume of liquid remains
 constant as the pool shape evolves.



metal plates:
 thickness is greater than the
 maximum melt zone depth, r
 width of gap g varies
 $length l \gg r$



Notes: Only liquid is shown. Total volume of
 liquid decreases as gap increases.

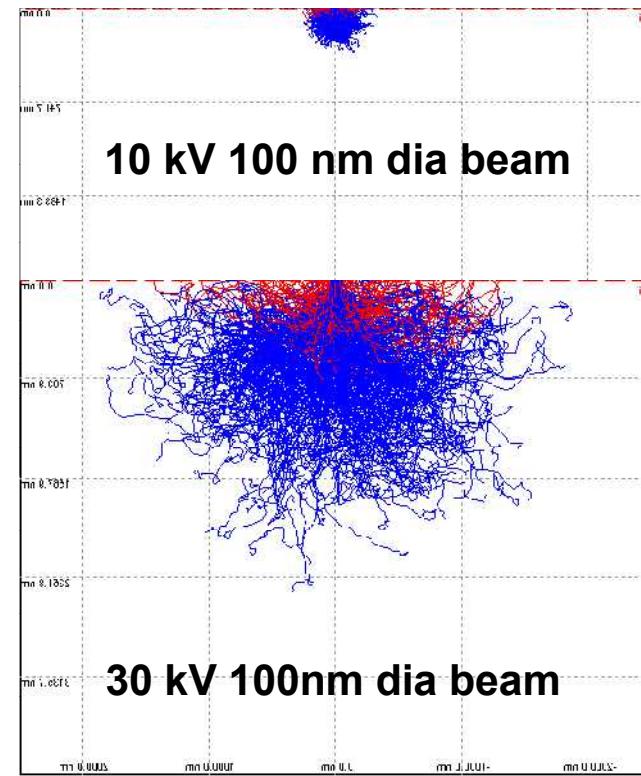
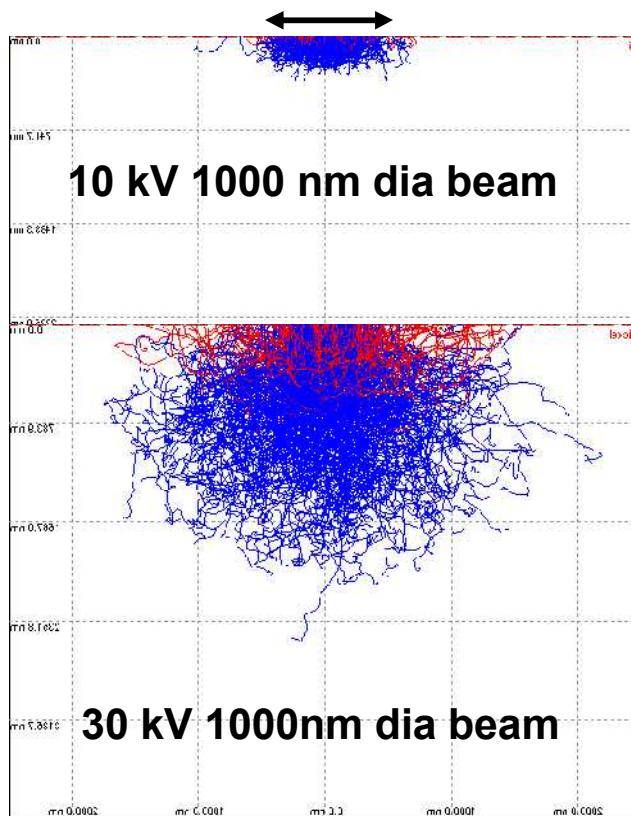


Beam Intensity: Electron Beams vs Lasers

	Beam Dia.	Beam Current	Accel. Voltage	Power	Intensity (kW / mm ²)
Laser μ welder:	15 μ m	n.a.	n.a.	20 W	\sim 100
Standard eB welder:	\sim 0.5 mm	\sim 25 mA	\sim 150 kV	\sim 4 kW	\sim 20
SEM: (analysis)	600 nm	1 μ A	30 kV	\sim 30 mW	\sim 100 *
SEM: (imaging)	60 nm	1 pA	30 kV	30 μ W	\sim 10 *
Tweaked SEM:	5 μ m	\sim 25 μ A	30 kV	\sim 750 mW	\sim 40 *

e-Beam Intensity

When an SEM builder says, ".....the beam is X nm in diameter.....",
What does that really mean for energy deposition?



Can't directly compare lasers vs electron beam surface intensities



Beam Comparison: Laser vs e⁻-Beam

Equation for radiation absorption by a medium:

$$I = I_0 \exp[-\mu z]$$

Energy Absorbed vs depth /Area:

$$E(z) = \int_0^z I_0 \exp[-\mu z] dz$$

Assume laser and E beam have same power and beam size:

$$E_{\text{laser}} = \int_0^{\infty} I_{\text{laser}} \exp[-\mu_{\text{laser}} z] dz = E_{\text{eb}} = \int_0^{\infty} I_{\text{eb}} \exp[-\mu_{\text{eb}} z] dz$$

which can be reduced to:

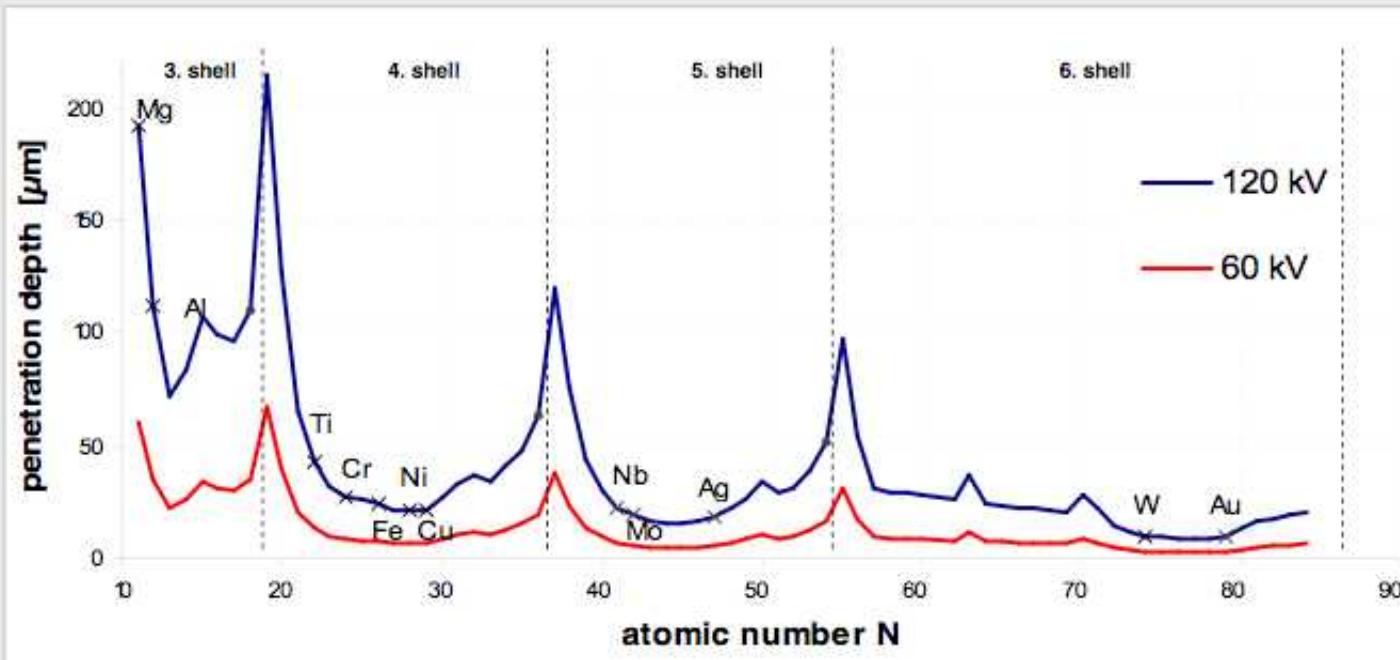
$$I_{\text{laser}} / I_{\text{eb}} = \mu_{\text{laser}} / \mu_{\text{eb}}$$

For Ni, $\mu_{\text{laser}} = 50 \text{ } \mu\text{m}^{-1}$, $\mu_{\text{eb}} = 2 \text{ } \mu\text{m}^{-1}$ @ 30 keV

thus, the E beam's surface intensity is effectively 25x lower
than an equivalent laser beam

Material / e^- -Beam Interaction

Electron penetration depth versus atomic number

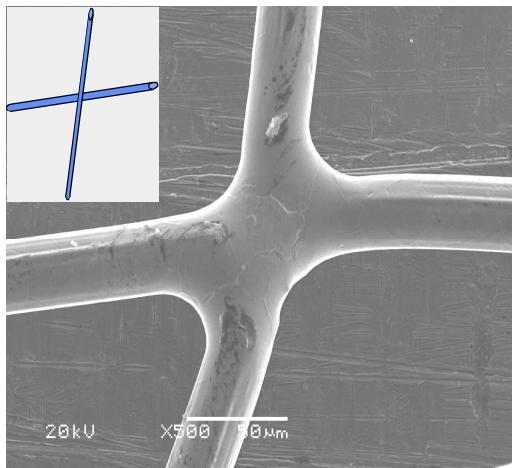


© pro-beam / 71

Thanks to Thorsten Löwer, ProBeam AG & Co



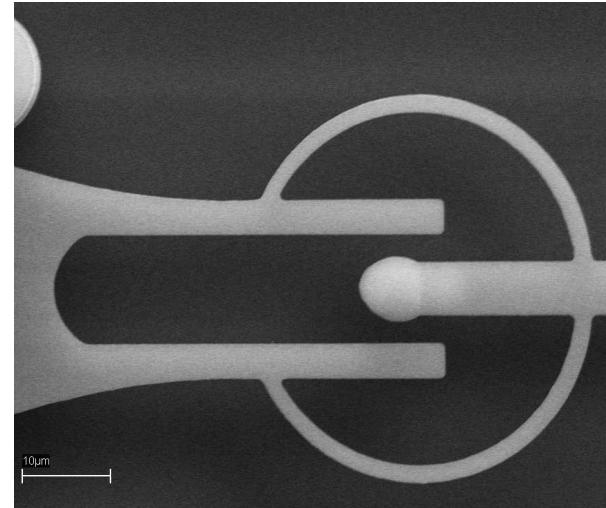
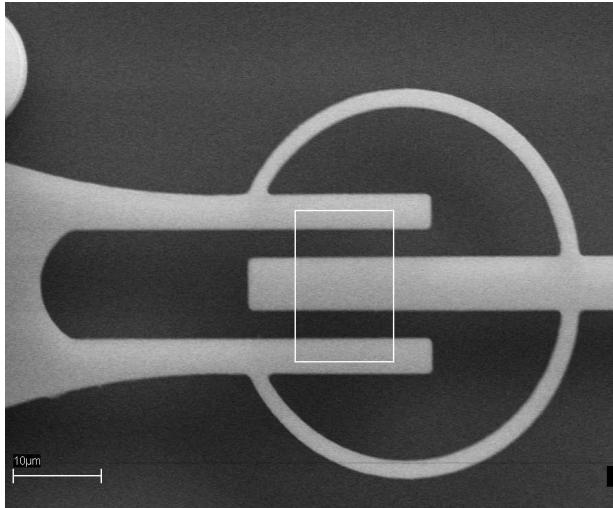
Examples



Cross-wire weld

Tophet C

30 μm diameter



**Unsuccessful weld
of poly-Si MEMS
device; did not
bridge gap**

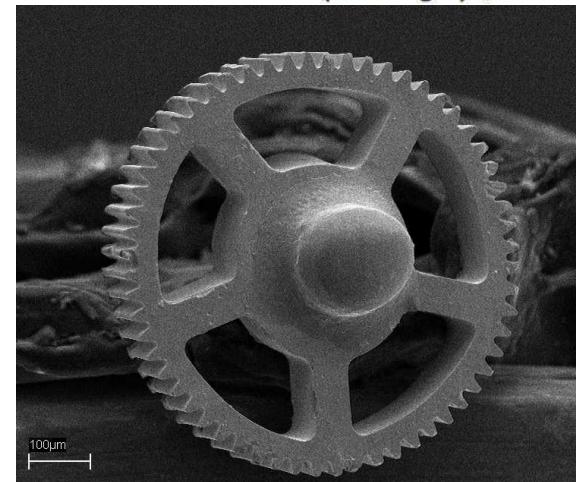
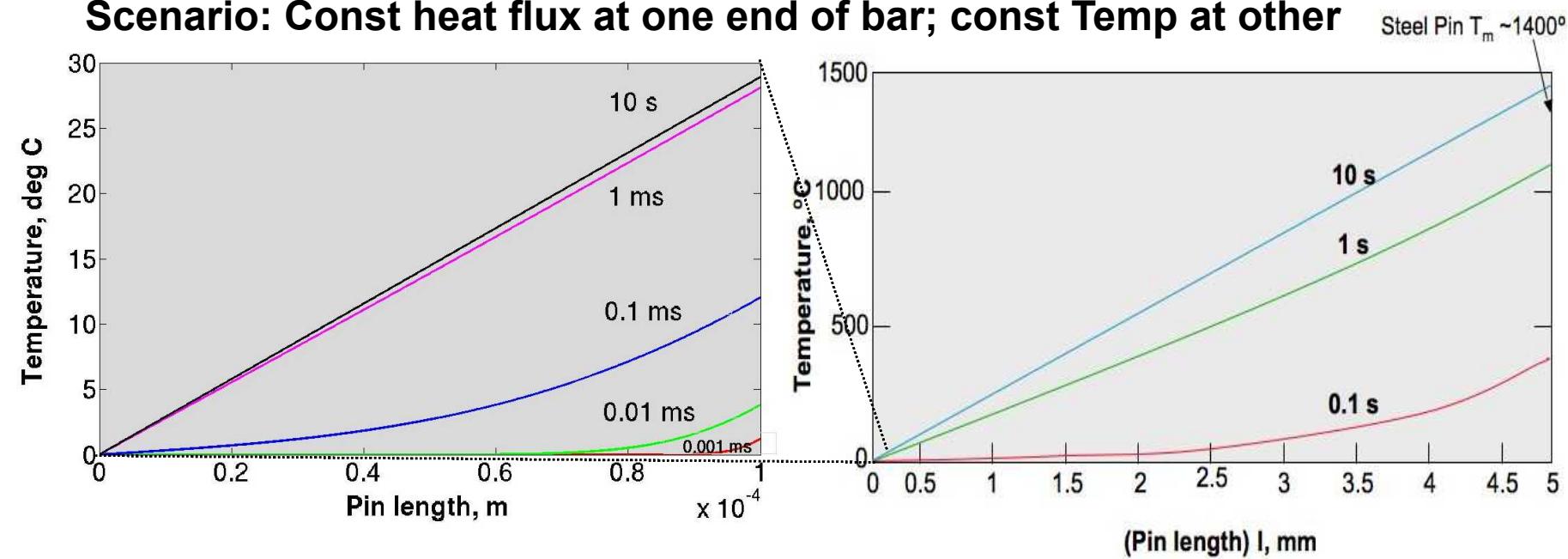


Thermal Considerations

- If heat sink is nearby
 - Temperature distribution rapidly achieves steady-state
 - Thermal influence distance: $TID \sim (\alpha t)^{1/2}$
 - Pick $\alpha \sim 0.1 \text{ cm}^2/\text{s} = 10^7 \mu\text{m}^2/\text{s}$ (value for steel)
 - TID of $100 \mu\text{m}$ implies $t \sim 1 \text{ ms}$
 - TID of $10 \mu\text{m}$ implies $t \sim 10 \mu\text{s}$
 - Considerable heat input needed if material being welded is a good conductor.
- If no heat sink is nearby
 - System acts like adiabatic block
 - Temperature rises very quickly and uniformly.
 - $7.2 \text{nJ}/(\mu\text{m})^3$ raises Si from ambient to melt. An "adiabatic" MEMS feature $100\mu\text{m} \times 10\mu\text{m} \times 1\mu\text{m}$, needs only 7.2 mJ, applied in $\sim 10 \text{ ms}$ by tweaked SEM.

Thermal Considerations: Heat Sink

Scenario: Const heat flux at one end of bar; const Temp at other





No Heat Sink





Examples of μ e-Beam welds

from U. Reisgen, T. Dorfmüller, ISF Aachen Univ.

QuickTime™ and a
TIFF (LZW) decompressor
are needed to see this picture.

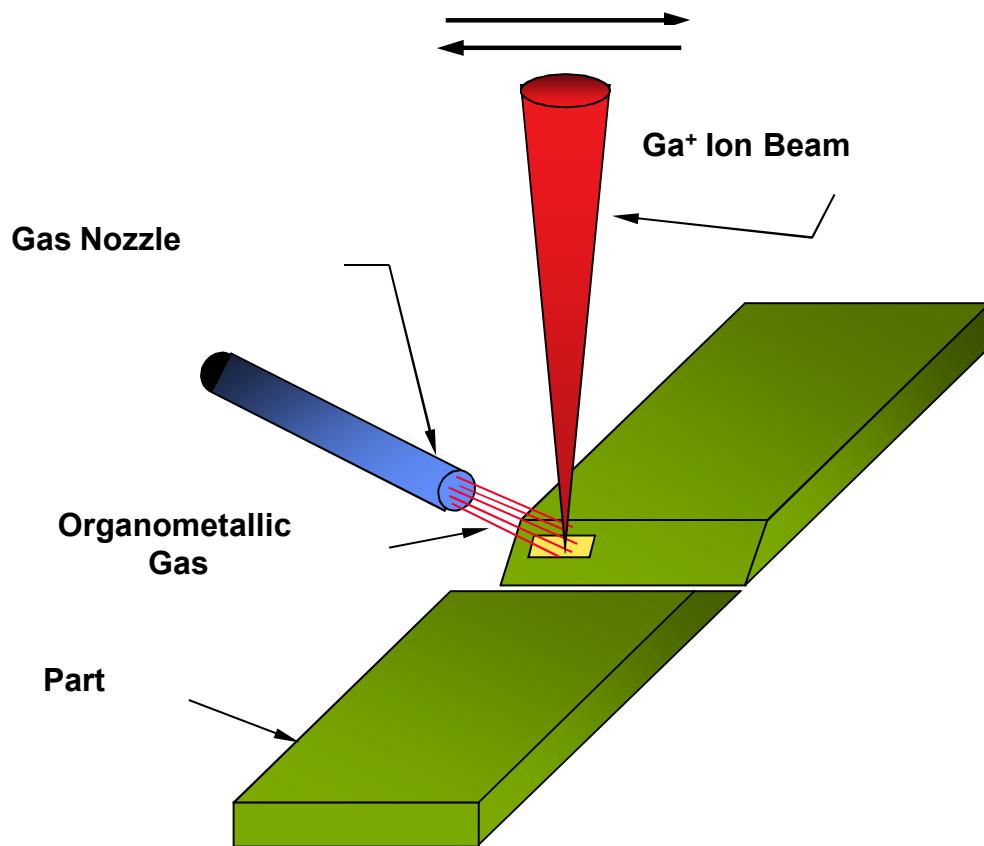
QuickTime™ and a
TIFF (LZW) decompressor
are needed to see this picture.

Axial butt and cross wire welds (175 μ m dia. Ck101)

QuickTime™ and a
TIFF (LZW) decompressor
are needed to see this picture.

QuickTime™ and a
TIFF (LZW) decompressor
are needed to see this picture.

Focused Ion Beam Joining



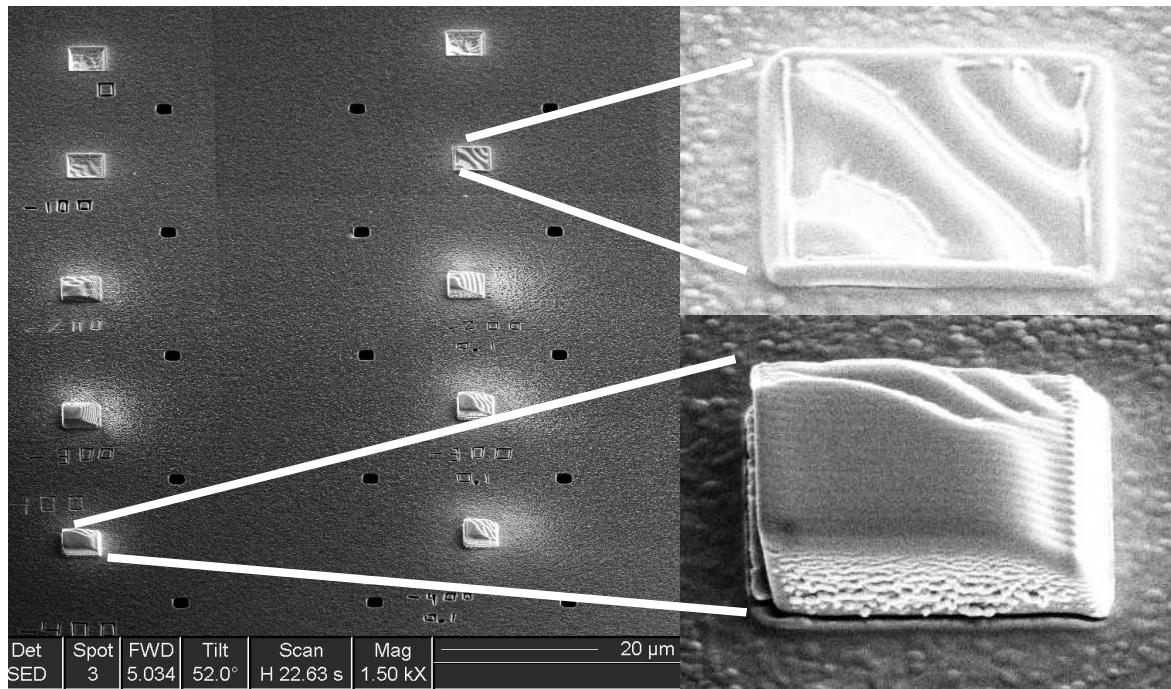
Ion beam "cracks" organometallic gas which has adsorbed onto surface

Can also be done with electron beam, but slower.



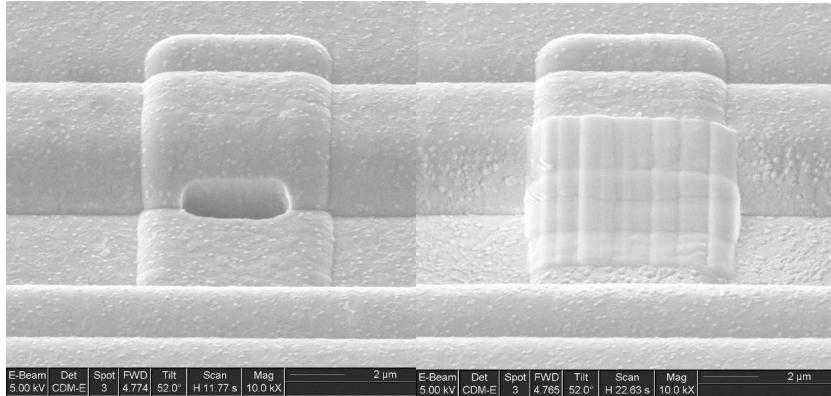
Characterizing the FIB process

FIB deposit as function of ion beam parameters: dwell and overlap
Observe gas flow eddies

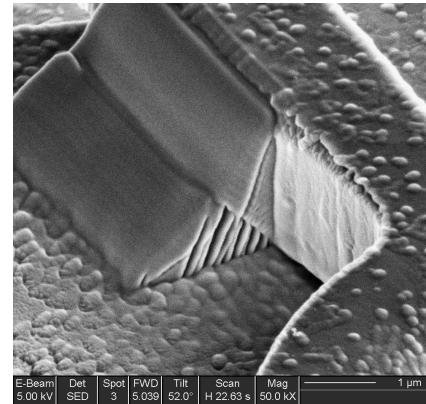




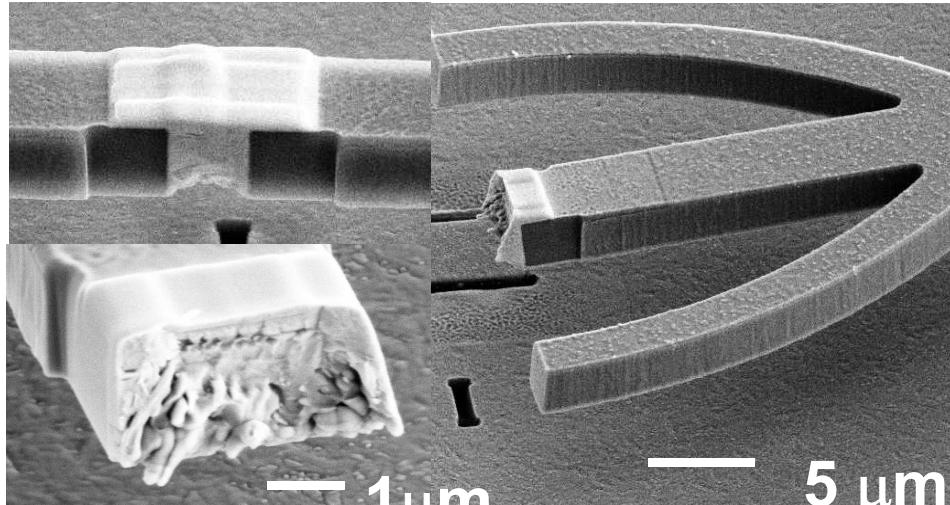
Example Joints



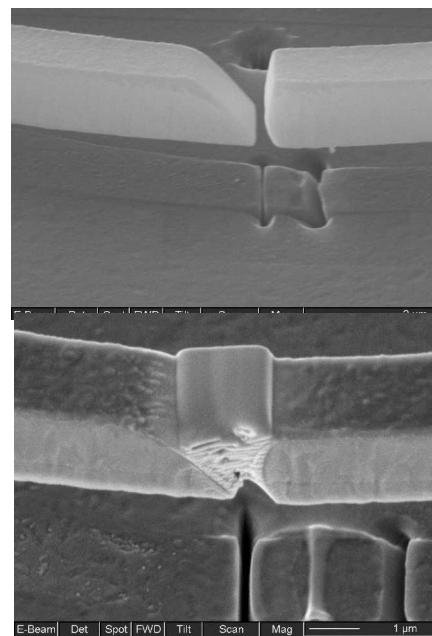
Etch hole patch in MEMS pump



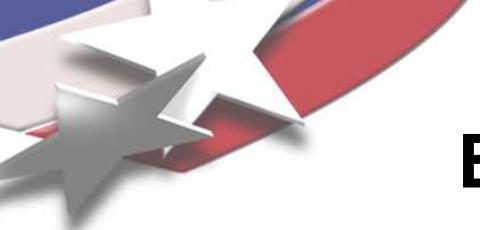
**Fillet joint in
MEMS test
bar**



"wide gap" joint in MEMS test bar



**"cut and
rejoin
narrow gap"
joint in
MEMS test
bar**



Exothermic Nano Laminates

Technique uses nanoengineered foil comprised of many thin alternating layers of materials with exothermic heat of mixing (e.g. Al-Ni) to provide controllable in-situ intense heat source.

re.

QuickTime™ and a
TIFF (LZW) decompressor
are needed to see this picture.

QuickTime™ and a
TIFF (LZW) decompressor
are needed to see this picture.

from:

**T Rude, D Van Heerden, TP Weihs, OM Knio, Reactive NanoTechnologies, Inc.
MT Powers, CD Enns, Agilent Technologies**



Summary

- The small size and delicate nature of microparts makes process observation and handling major difficulties.
- Success requires careful attention to: energy input (amount and location), part fitup, fixturing and surface effects (especially for fusion processes).
- Nevertheless, many processes can be used to create successful microjoints, in both spot and seam geometries (though spot welds are predominant, by far)



Future Developments?

- Self-assembling molecules making microjoining a wet-chemical massively parallel process (already used to produce coatings for tribological control)
- Tailored nano-particles that aggregate and react at preferred locations
- Joining of nanotubes/nanowires will become important in microdevices
- Photon (or other energy form) stimulated bonding reactions (like today's UV-cured adhesives)